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Recent advances in transition-metal-sulfide-based bifunctional electrocatalysts for overall water splitting

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Hydrogen produced *via* water electrolysis can act as an ideal clean chemical fuel with superb gravimetric energy density and high energy conversion efficiency, solving the problems of conventional fossil fuel exhaustion and environmental contamination. Transition metal sulfides (TMS) have been extensively explored as effective, widely available alternatives to precious metals in overall water splitting. Herein, recent advances, covering preparation methods, intrinsic electrocatalytic performance, and optimization strategies, relating to TMS-based bifunctional electrocatalysts have been summarized systematically and comprehensively. Firstly, a general introduction to the reaction mechanisms and key parameters of the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) is provided. Next, the physicochemical properties of TMS and typical synthesis methods are introduced to give guidance for fabricating TMS materials with well-defined structures, controllable compositions, and excellent performance. Importantly, the intrinsic activities of TMS-based electrocatalysts and several strategies for improving their bifunctional electrocatalytic performance during water electrolysis are discussed in detail. Finally, perspectives covering the challenges and opportunities related to the further development of TMS-based materials with high activity and long-term durability for overall water splitting are given. The aim herein is to provide guidelines for the design and fabrication of TMS-based bifunctional electrocatalysts with excellent performance and to accelerate their large-scale practical application in water electrolysis.

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1. Introduction

1.1 Background

The growing consumption of conventional fossil fuels and accompanying environmental contamination are affecting global society on an unprecedented scale.¹ Consequently, there



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is a quite imminent need to fundamentally adjust the world energy landscape and build clean, low-carbon, safe, and efficient modern energy systems. As an ideal clean chemical fuel with superb gravimetric energy density and energy conversion efficiency, hydrogen energy is expected to be an excellent candidate to replace traditional fossil fuels.² Electrochemical water splitting is considered to be one of the most promising hydrogen production technologies, and it can utilize electricity generated *via* renewable energy sources, such as solar energy, wind power, geothermal energy, and bioenergy, to form a closed loop of renewable energy.³ Unfortunately, the large-scale commercial application of electrochemical water splitting is subject to the following three restrictions: (i) a larger overpotential than the theoretical value (1.23 V) being needed to drive overall water splitting; (ii) the poor stability of electrode materials; and (iii) high cost caused by the scarcity of noble-metal electrocatalysts.^{4,5} To date, the benchmark electrocatalysts for the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) are Pt-group- and Ru/Ir-based noble metal materials with small overpotentials and low Tafel slopes.⁶ These noble metals can suffer from dissolution, agglomeration, and poor durability during the water splitting process.⁷

The development of low-cost and widely available alternatives with more excellent activities and stabilities than precious metals is urgently needed for overall water splitting, and this is a crucial step in the development of a hydrogen-based economy. There are three main kinds of electrolyzer used for water electrolysis, categorized by the electrolyte: alkaline electrolyzers (AEs), acidic proton exchange membrane (PEM) electrolyzers, and solid oxide electrolyzers (SOEs).⁸ SOEs are still in the laboratory stage, and they are operated at high temperatures (700–800 °C) to reduce the overpotential. Alkaline electrolyzers have the advantages of simple construction, convenient operation, and low cost. In comparison, water electrolysis carried out in acidic electrolyte involves fewer adverse reactions and higher ionic conductivity, but it demands expensive perfluorinated Nafion-based PEMs.⁹ As we know, neutral and alkaline electrolytes are favorable for the OER, while acidic electrolytes are beneficial for the HER. Inevitably, this non-negligible mismatch

between optimal working environments for the HER and OER has a negative impact on the overall efficiency of water electrolysis systems.¹⁰ In this regard, designing and constructing effective bifunctional electrocatalysts that can be employed for both the HER and OER in the same electrolyte is another vital issue related to commercial electrochemical water splitting.

A variety of Earth-abundant and non-precious transition metal (*e.g.*, Mo, W, Ni, Co, and Fe) based compounds have been explored as bifunctional electrocatalysts for overall water splitting, such as transition metal sulfides, phosphides, nitrides, carbides, oxides, and hydroxides.^{3,10–14} Specifically, transition-metal sulfides (TMS) with distinctive structural features, rich active sites, and adjustable electronic properties and components have attracted widespread research attention,² *e.g.*, layered MoS₂ and WS₂ and non-layered Ni₃S₂ and Co₃S₄. There are multitudinous excellent reviews on research achievements relating to TMS-based electrocatalysts, however, most of them have an emphasis on either one or a limited number of TMS systems for electrochemical water splitting, such as transition-metal dichalcogenides for the HER,^{15,16} Fe-based electrocatalysts for the OER,¹⁷ or Co-based electrocatalysts for overall water splitting.¹⁸ Moreover, the use of nanoarchitectonics in TMS-based electrocatalysts to improve their performance for water splitting¹⁹ and recent advances relating to metal sulfides, from their controlled fabrication to their applications in electrocatalytic, photocatalytic, and photoelectrochemical water splitting, have also been reviewed.² However, a targeted review focusing on TMS-based bifunctional electrocatalysts addressing their controllable preparation, intrinsic electrocatalytic activities, and optimization strategies for enhancing their electrocatalytic performance in overall water splitting has not been fully provided to date.

Herein, we focus on recent developments relating to TMS-based bifunctional electrocatalysts, with respect to synthetic methods, intrinsic electrocatalytic activities, and corresponding optimization strategies, aiming to provide a comprehensive overview of the basic information and breakthroughs relating to the use of TMS for electrochemical water splitting. Firstly, the reaction mechanisms for the HER and OER processes together



Fig. 1 Possible suitable elements and strategies for improving the electrocatalytic performances of TMS-based bifunctional electrocatalysts; transition metals with deeper colors in the periodic table have been the focus of more studies.

with some significant parameters for evaluating the properties of bifunctional electrocatalysts are introduced. Secondly, the physicochemical properties of TMS and the role of S atoms in electrochemical water splitting are emphatically discussed. Then, typical synthetic methods for TMS are discussed, which exhibit great influence on the electrocatalytic performance. Importantly, the intrinsic activities of TMS-based bifunctional electrocatalysts and several strategies aimed at the optimization of the electrocatalytic performance in overall water splitting are summarized systematically. Possible suitable elements and strategies for improving the electrocatalytic performance of TMS-based bifunctional electrocatalysts are displayed in Fig. 1. Meanwhile, the opportunities and challenges relating to the development of TMS bifunctional electrocatalysts are discussed and an outlook shedding light on directions for further research into TMS is also put forward.

1.2 Fundamentals of overall water splitting

1.2.1 Reaction mechanisms. Electrochemical water splitting involves two half reactions, namely the HER and OER. The electrolyzer typically consists of three parts: a cathode, an anode, and the aqueous electrolyte, and a schematic diagram is shown in Fig. 2a. When a certain voltage is applied to the two electrodes, the water reduction reaction occurs at the cathode to produce H_2 and the water oxidation reaction happens at the anode to produce O_2 . The total reaction can be written as follows:



The theoretical thermodynamic potential of overall water splitting is 1.23 V under ideal conditions (25 °C and 1 atm). However, a higher operational potential (E) is always necessary to actually carry out the process. The excess potential is named the overpotential (η), which is mainly used to overcome intrinsic activation hindrance at both the anode (η_a) and cathode (η_c), as well as other inevitable obstacles (η_{other}) due to the electrolyte and contact resistance. Overall:

$$E = 1.23 \text{ V} + \eta_a + \eta_c + \eta_{other} \quad (2)$$

The HER is a two-electron transfer process that happens at the cathode; it is always more active in acidic electrolytes than in alkaline electrolytes because the reduction of protons to H_2 is energetically more favorable.²⁰ There are two mechanisms involved the HER, namely the Volmer–Heyrovsky and Volmer–Tafel pathways (Fig. 2b), depending on the way in which adsorbed hydrogen is desorbed from the electrocatalyst.²¹ In acidic electrolytes, hydronium ions adsorb on the cathode to form hydrogen intermediates (H^*) *via* the Volmer reaction ($H_3O^+ + e^- \rightarrow H^* + H_2O$) and then transform into H_2 *via* the Heyrovsky reaction ($H^* + H_3O^+ + e^- \rightarrow H_2 + H_2O$) at low H^* coverage or *via* the Tafel reaction ($H^* + H^* \rightarrow H_2$) at high H^* coverage. Under alkaline conditions, the HER is more sluggish due to the water dissociation step prior to the formation of H^* .²² A water molecule undergoes a dissociation reaction to generate adsorbed H^* *via* the Volmer reaction ($H_2O + e^- \rightarrow H^* + OH^-$). The desorption process is carried out *via* the Heyrovsky or Tafel reaction, similar to under acidic conditions.

In the HER process, the free energy of hydrogen adsorption (ΔG_H) calculated *via* density functional theory (DFT) is widely accepted as a descriptor for the interactions between adsorbed H^* and the electrocatalyst.²³ A large negative ΔG_H value indicates that the adsorption of hydrogen on the electrode is much easier than desorption, so the Heyrovsky or Tafel reaction is the rate-limiting step. Otherwise, a large positive ΔG_H value represents a weak interaction, and the rate-limiting step is the Volmer reaction. Thus, an excellent HER electrocatalyst should possess near-zero ΔG_H , with a balance between the absorption and desorption of hydrogen.

The OER is more sluggish than the HER, and it naturally requires a larger overpotential because it involves a complex four-proton transfer process.¹⁹ Possible mechanisms for the OER at the anode have been proposed, including almost the same intermediates, such as MO and MOH (M: transitional metal, such as Fe, Co, and Ni).²⁴ The general reaction pathways and widely accepted mechanisms for the OER in alkaline and acidic electrolytes are shown in Fig. 2c. There are two possible pathways for MO intermediates to convert to O_2 : combining H_2O under acidic conditions or OH^- under alkaline conditions with an MO intermediate to form an MOOH intermediate, which then transforms into O_2 ; and directly uniting two MO



Fig. 2 (a) A schematic illustration of electrochemical water splitting. (b) The general reaction pathways for the HER in alkaline (red lines) and acidic (blue lines) electrolytes. (c) The general reaction pathways for the OER in alkaline (red lines) and acidic (blue lines) electrolytes; the black and purple lines stand for two possible intermediates: M–OOH and M–O, respectively.

intermediates to generate O₂. Therefore, the M–O bonds of MOH, MO, and MOOH intermediates play key roles in determining the electrocatalytic OER activity.²⁵ Since the reactions in the OER are all thermodynamically uphill, the reaction with the highest energy barrier will be the rate-determining step.¹⁹ The OER rate-determining step can also be determined based on the Tafel slope, and the actual reaction mechanism can be understood further. For example, the Tafel slope is about 120 mV dec⁻¹ when the first reaction (from M to MOH) is the rate-determining step, while the Tafel slope is about 40 mV dec⁻¹ if the second reaction (from MOH to MO) acts as the rate-determining step.

The thermodynamics in each OER step can be studied *via* DFT calculations. Rossmeisl *et al.* studied four reactions during the OER and obtained the corresponding Gibbs free energies ($\Delta G_i = \Delta G_{\text{product}} - \Delta G_{\text{reactant}} - eU - k_B T \ln[\text{H}^+]$, $i = 1-4$) under an applied potential (U), where k_B is the Boltzmann constant and T is the temperature.²⁶ The largest Gibbs free energy of these is defined as ΔG^{OER} for the OER.³ Under standard conditions ($U_{\text{SHE}} = 0$), the relationship between the theoretical overpotential (η^{OER}) and ΔG^{OER} is as follows:

$$\eta^{\text{OER}} = (\Delta G^{\text{OER}}/e) - 1.23 \text{ V} \quad (3)$$

The ideal value of ΔG^{OER} is 1.23 eV when $\eta^{\text{OER}} = 0$, indicating that the Gibbs free energy in each step is 1.23 eV. However, there is always a gap between ideality and reality due to unsuitable electrocatalyst oxygen bonding forces that can make the OER more sluggish.

1.2.2 Parameters for electrochemical water splitting. There are some important electrochemical water splitting parameters that can be used to evaluate the electrocatalytic performances of materials, such as the overpotential, Tafel slope, exchange current density, turnover frequency, stability, faradaic efficiency, and electrochemically active surface area.

Overpotential (η) is the excess part of the operational potential beyond the equilibrium potential that arises due to an unavoidable intrinsic kinetic barrier.²⁷ As mentioned above, the equilibrium potentials for electrocatalyzing the HER and OER are 0 V and 1.23 V (*vs.* RHE: reversible hydrogen electrode), respectively. However, applied potentials much higher than the equilibrium potentials are needed to overcome the electrode kinetic barriers in the reaction. Normally, the total activity of an electrocatalyst can be assessed based on polarization curves (after IR compensation) using the geometric current density and overpotential, which are obtained from cyclic voltammetry (CV) or linear sweep voltammetry (LSV) measurements. It is worth noting that different current densities can refer to different overpotential values. Thus, the overpotential at a certain current density (like 10 or 100 mA cm⁻²) obtained from a polarization curve is one of the most popular descriptors of the total activity.²⁸ The smaller the overpotential at a certain current density, the better the electrochemical activity of the electrocatalyst.

Tafel slope is a key kinetic parameter in electrochemical water splitting, implying how fast the current density increases with increasing overpotential.²⁹ A Tafel plot can be obtained *via*

transforming the current density from a polarization curve into a logarithmic value with base 10 on the x axis, with the corresponding overpotential on the y axis. In a Tafel plot, there is a linear region that can be well-fitted using the Tafel equation:

$$\eta = a + b \log j \quad (4)$$

where b stands for the Tafel slope, η is the overpotential, j represents the current density, and a is a constant. A smaller Tafel slope always signifies faster charge-transfer kinetics; in other words, increasing the current density leads to a smaller overpotential increase than in a case with a larger Tafel slope. Hence, the Tafel slope can be employed to deduce the possible reaction mechanism, and especially to explain the rate-determining step. When the value of η is equal to zero, the corresponding current density (j) is the exchange current density (j_0) that can reflect the intrinsic bonding/charge transfer interactions between the electrocatalyst and reactant. A high j_0 value is always a good indication of superior inherent electrocatalyst activity under equilibrium conditions. In short, materials with a small η and a larger j_0 value will be efficacious electrocatalysts.

Turnover frequency (TOF) denotes the number of molecules transformed at each active site per unit time, signifying the intrinsic activity of each active site. The TOF can be calculated *via* the following equation:

$$\text{TOF} = (jA)/(\alpha Fn) \quad (5)$$

where j represents the current density at a certain overpotential obtained from the LSV curve; A is the surface area of the electrode; α denotes the electron number of the electrocatalyst (electrons mol⁻¹); F is the Faraday constant (96 485.3 C mol⁻¹); and n stands for the molar amount of electrocatalyst. The precise TOF value is difficult to obtain, as it is not possible to accurately determine the number of all accessible active sites involved in the actual reaction. As such, a reasonable approach for calculating the TOF is based merely on either the number of atoms on the surface or the number of easily accessible electrocatalytic sites of the material. Sometimes, the TOF is calculated based on the overall catalytic species existing in the material.¹¹ Despite the imprecision of TOF values, they still play a meaningful role in evaluating the intrinsic catalytic activities of electrocatalysts with similar elements and structures.

Stability is a considerably important parameter for electrocatalysts for practical applications, reflecting the ability of an electrocatalyst to maintain its activity under long-term testing (generally several or tens of hours). In general, there are two electrochemical techniques used to measure the electrocatalytic stability. One is performing CV measurements for thousands of cycles, followed by LSV measurements; the stability can be examined *via* comparing the LSV curves before and after the cycling CV testing. The other way is to conduct chronoamperometry or chronopotentiometry measurements to show how the potential or current density varies with time; a high retention rate corresponds to good stability. Usually, a current density of 10 mA cm⁻² is applied as a standard for

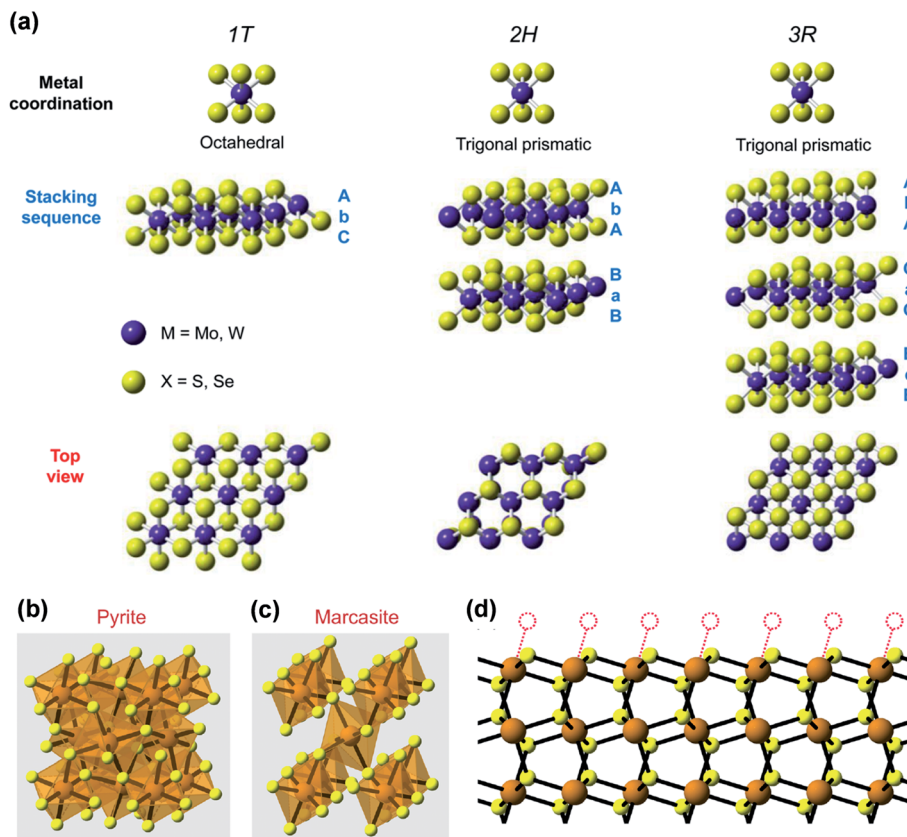


Fig. 3 (a) The different metal coordination modes and stacking sequences of layered TMS structural unit cells. The metal can have either octahedral coordination or trigonal prismatic coordination. Octahedral coordination allows stacking sequences with tetragonal symmetry, like AbC, AbC, etc.; trigonal prismatic layers can be stacked in two different ways to form hexagonal symmetry (2H) or rhombohedral symmetry (3R).³⁴ (b and c) The structures of non-layered TMS in pyrite and marcasite phases, in which transition metal atoms and S atoms are shown in orange and yellow, respectively. (d) A side-view of the stable nonpolar pyrite (100) facet as an example of a low-index surface with under-coordinated transition metal cations.³⁸

evaluating the electrocatalytic stability. To illustrate the potential of an electrocatalyst for use in practical applications, stability at a large current density (e.g., 500 mA cm^{-2}) is absolutely needed.

Faradaic efficiency is defined as the efficiency of electron transfer when catalyzing a desired reaction in the electrochemical system, which is quantified as the experimental to theoretical molar ratio of the numbers of gas molecules. The experimental produced gas can be measured *via* the traditional water gas displacement method or gas chromatography, while the theoretical value can be calculated *via* chronoamperometry or chronopotentiometry measurements. A higher faradaic efficiency means less energy loss during the electrochemical reaction.

Electrochemically active surface area (ECSA) is universally utilized to estimate the effective active surface of the working electrode, which is based on the electrochemical double-layer capacitance of the electrocatalytic surface.³⁰ The electrochemical double-layer capacitance can be measured based on the capacitive current associated with double-layer charging in a non-faradaic region from the scan-rate dependence of CV curves.³¹ Typically, the non-faradaic region is a potential

window of 0.1 V centered at the open-circuit potential of the system. A plot of the charging current as a function of the scan rate yields a straight line with a slope equal to the electrochemical double-layer capacitance (C_{DL}). The ECSA of an electrocatalyst sample is calculated *via* the following equation:

$$\text{ECSA} = \frac{C_{DL}}{C_s} \quad (6)$$

where C_s is the specific capacitance of the electrocatalyst or the capacitance per unit area of an atomically smooth planar surface of the material, with typical values between 0.015 and 0.110 mF cm^{-2} in H_2SO_4 and 0.022 and 0.130 mF cm^{-2} in NaOH or KOH solution.³⁰ In addition, the electrochemical capacitance can also be acquired from the frequency-dependent impedance based on electrochemical impedance spectroscopy (EIS) studies in the same non-faradaic region.

2. Physicochemical properties of TMS

2.1 Structures and basic properties

Based on crystal structures, TMS can be mainly divided into two categories: layered MS_2 and non-layered M_xS_y . The sulfides of transition-metals in groups 4–7 ($\text{M} = \text{Mo}, \text{W}, \text{Ta}, \text{Nb}, \text{etc.}$) always

crystallize with a graphite-like layered structure, whereas some sulfides of transition-metals in groups 8–12 ($M = \text{Fe}, \text{Co}, \text{Ni}, \text{etc.}$) predominantly adopt non-layered structures. Distinctions between the structures and compositions of these TMS will inevitably bring about differences in their electronic and electrochemical properties.

Typically, each layer of layered MS_2 has a thickness of 0.6–0.7 nm, consisting of a hexagonal layer of transition-metal atoms sandwiched between two layers of sulfur (S) atoms.³² These layers, with strong intra-layer covalent bonds between atoms, will be stacked vertically together *via* relatively weak out-of-plane *van der Waals* forces. Consequently, layered MS_2 can be exfoliated readily into two-dimensional nanosheets.³³ Depending on the transition-metal coordination of S atoms and the stacking sequence adopted by multiple layers, layered MS_2 can be classified into different phases: 1T, 2H, and 3R, where the digit denotes the number of stacked layers in the crystallographic unit cell and the letter stands for tetragonal, hexagonal, and trigonal lattices, respectively, as shown in Fig. 3a.³⁴ In layered MS_2 , the transition-metal atoms centered in an octahedral configuration or trigonal prismatic configuration will provide four electrons to fill the bonding states of TMS, so the transition-metal atom is in a +4 oxidation state and the S atom is in a –2 state.

The electronic properties of TMS are extremely dependent on the filling of the d orbitals of the transition metal. For example, the semiconductor behavior of natural 2H-MoS₂ can mainly be ascribed to the fully filled d orbitals of the Mo atom, so 2H-MoS₂ is expected to be employed in electronic devices. However, 1T-MoS₂, with partially filled Mo atom d orbitals exhibits metallic properties, which will be more suitable for electrocatalysis.³⁵ Interestingly, an MS_2 phase transition from 2H to 1T, such as in the cases of MoS₂ and WS₂, could be realized *via* the intercalation of alkali metals, which introduce extra electrons and rearrange the d orbitals of metals.^{36,37}

Furthermore, the electrochemical properties of layered TMS also strongly depend on their structures and compositions. The unique stacked structures of layered TMS bring about two distinctive orientations: the basal plane and edge plane, which exhibit anisotropic properties in many aspects. Firstly, the in-plane electrical conductivity of layered MS_2 is about 2200 times higher than the interlayer conductivity.³⁴ As a consequence, heterogeneous electron transfer along the edge plane of layered TMS is significantly faster than along its basal plane. As such, the electrochemical activity in the edge plane and basal plane of layered TMS is also predicted to be anisotropic. Typically, sites on the basal planes of group-6 TMS are always inert, while the highly active edge sites mainly contribute to the excellent electrocatalytic performance.³⁹ Taking well-studied MoS₂ as an example, 2H-MoS₂ exhibits hydrogen adsorption free energy of +0.08 eV on its edge sites compared with 2.00 eV on its inert basal plane.⁴⁰ Intensive efforts have been made to induce basal plane electrochemical activity. Zheng *et al.* activated and optimized the basal plane of monolayer 2H-MoS₂ for the HER *via* introducing S vacancies and elastic strain.⁴¹ S vacancies could serve as new active sites in the basal plane, as the gap states around the Fermi level would permit hydrogen to

bind with exposed Mo atoms directly. What is more, the electrochemical activity of 2H-MoS₂ could be optimized further *via* straining the plane with S-vacancy sites.

In addition, chemical doping with non-metallic or metallic elements is widely considered to be an effective approach for creating active sites in the inert basal plane of MoS₂ and for increasing the intrinsic electrical conductivity; *e.g.*, oxygen (O),⁴² nitrogen (N), phosphorus (P),⁴³ selenium (Se),⁴⁴ and zinc (Zn)⁴⁵ elements can be used. Additionally, these dopants can also greatly optimize the hydrogen adsorption free energy, accelerate charge transfer, and, finally, enhance the electrocatalytic performance of MoS₂. It is worth mentioning that group-5 TMS can possess active basal-plane sites intrinsically, *e.g.*, in the cases of TaS₂ and NbS₂. Jakobson *et al.* proposed that active sites were concentrated in the basal planes and at the edges of H-TaS₂ simultaneously according to experimental and theoretical results, and the active basal-plane sites could contribute to self-optimizing behavior during the HER.⁴⁶

Non-layered M_xS_y ($M = \text{Fe}, \text{Co}, \text{Ni}, \text{etc.}$) generally shares a pyrite structure or marcasite structure when $x = 1$ and $y = 2$, and the conventional unit cells with high similarity are shown in Fig. 3b and c.³⁸ The pyrite structure belongs to the space group $Pa\bar{3}$, in which the transition-metal atoms are located at face-centered cubic (FCC) sites and bonded octahedrally to adjacent S atoms. Each S atom in the pyrite structure is tetrahedrally coordinated to three transition-metal atoms and one S atom, such that an S₂ dimer will also be formed.⁴⁷ The marcasite structure adopts the orthorhombic $Pnmm$ space group, where the body-centered transition-metal atoms are also octahedrally bonded to neighboring S atoms. It can be clearly found that the octahedra in the marcasite structure are edge-sharing, while in the pyrite structure they are corner-sharing. Indeed, the intergrowth or epitaxial growth of the marcasite structure in/on the pyrite structure of non-layered M_xS_y can be achieved readily due to their structural similarities. Moreover, the electronic structures of pyrite-type non-layered M_xS_y , which predominantly depend on the d-electron count of the transition metal, are diverse, ranging from insulators (such as NiS₂) to semiconductors (such as FeS₂) to metals (such as CoS₂).⁴⁸ Interestingly, the values of x and y for each particular M_xS_y structure can be diverse and a series of sulfides can be formed, *e.g.*, NiS, NiS₂, Ni₃S₂, Ni₃S₄, Ni₇S₆, and Ni₉S₈.⁴⁹ Typically, the composition of non-layered M_xS_y can also have a great influence on the electronic properties. In nickel sulfides, NiS₂ is an insulator while heazlewoodite Ni₃S₂ shows intrinsic metallic behavior owing to the continuous conductive network, connected *via* Ni–Ni bonds, in its crystal structure.⁵⁰

Differences in the crystal structures and compositions of layered MS_2 and non-layered M_xS_y result in discrepant electronic and electrochemical properties. However, the main strategies for improving the electrocatalytic performance are comprehensive, and they involve enhancing the intrinsic activity of each active site, increasing the number of active sites, and improving the electrical conductivity.^{1,51} Edge engineering and defect engineering have been developed to expose or create more active sites, and strain engineering, facet engineering, heteroatom doping, and composite designing have been

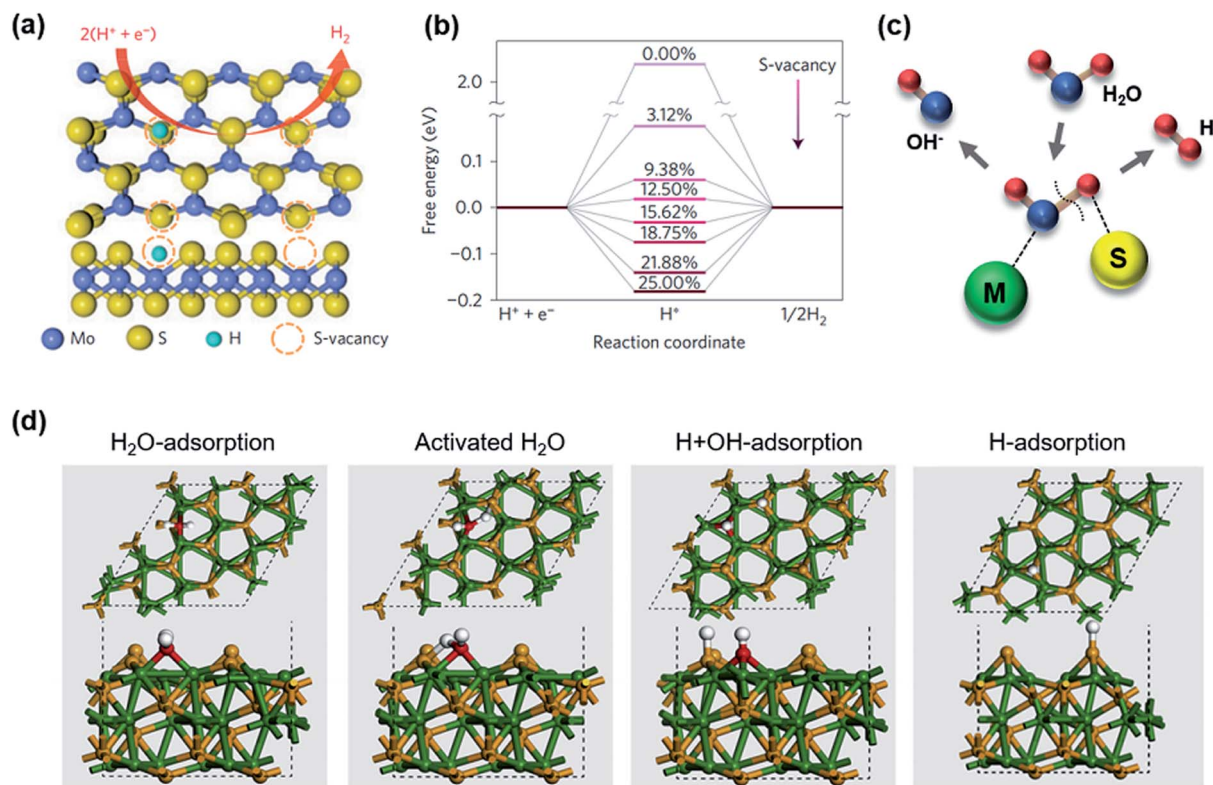


Fig. 4 (a) Schematic illustrations showing top and side views of MoS₂ with S vacancies on the basal plane. (b) A plot of calculated free energy versus the reaction coordinate of MoS₂ during the HER over an S vacancy range of 0–25%.⁴¹ (c) A schematic diagram of the HER reaction mechanism of TMS when metal atoms act as the active sites. (d) The obtained structures of H₂O, activated H₂O, OH, and H intermediates during water dissociation and H adsorbed on the Ni₃S₂ (003) surface after relaxation.⁵⁰

explored to accelerate electron transportation in these electrocatalysts during water splitting. Details of these optimization approaches for electrocatalytic performance will be discussed below.

2.2 The role of S atoms in the HER and OER

Revealing the true active sites or species is crucial for developing new electrocatalysts aimed at efficient overall water splitting. Although the main active sites in TMS are always metal atoms, understanding the role of S atoms in TMS is still an important task. Basically speaking, S atoms in TMS can contribute to special structures and abundant species, such as in the 2D layered structures of MS₂ (M: transition metal in group 4–7), and in the nickel sulfides Ni₃S₂, NiS₂, and Ni₃S₄. The roles of S atoms in catalyzing the HER and OER can be divided into two types: a direct role and an indirect role, depending on the contribution of S atoms to the electrocatalytic performance.

In the HER, some exposed S atoms and S vacancies on the edges or on the basal plane of TMS can have a great effect on the electrocatalytic performance; this can be thought of as an embodiment of the direct role of S atoms. Based on experimental and theoretical results, Jaramillo *et al.* identified that the active sites of MoS₂ during the HER could be S atoms on the edges under low H coverage (1/4 ML).⁵² What is more, S vacancies on the basal plane of layered 2H-MoS₂ could

contribute to enhanced HER activity,⁴¹ which is a powerful method for improving the HER performance (Fig. 4a). On one hand, S vacancies can change the coordination environment of adjacent metal atoms: metal atoms with low coordination will be more active and will possess higher intrinsic electrocatalytic activity. On the other hand, S vacancies can induce more active sites and significantly increase the number of active sites. Therefore, S vacancies can create new active sites on the inert basal plane of 2H-MoS₂, where gap states around the Fermi level can allow hydrogen binding directly with exposed Mo sites. Furthermore, ΔG_{H} can be finely manipulated *via* changing the number of S vacancies (Fig. 4b).

The indirect role of S atoms in the HER mainly arises through providing a place for hydrogen adhesion and separation when metal atoms act as the active sites (Fig. 4c). When the HER took place on Ni₃S₂ nanoporous thin films with exposed (003) facets, Dong *et al.* came to the following conclusions based on theoretical calculations: H₂O was preferentially adsorbed on Ni sites *via* the Volmer process, and S sites were more suitable for the formation of H₂ in the Tafel stage (Fig. 4d).⁵⁰ More straightforward evidences for the indirect role of S atoms has been found in experiments by Staszak-Jirkovskiy *et al.*⁴ They proposed that TMⁿ⁺ (Co²⁺/Mo⁴⁺) plays a key role in accelerating the sluggish water dissociation process in alkaline solutions, while the recombination of adsorbed hydrogen could happen at S sites. Due to the amorphous phases of the catalysts, they

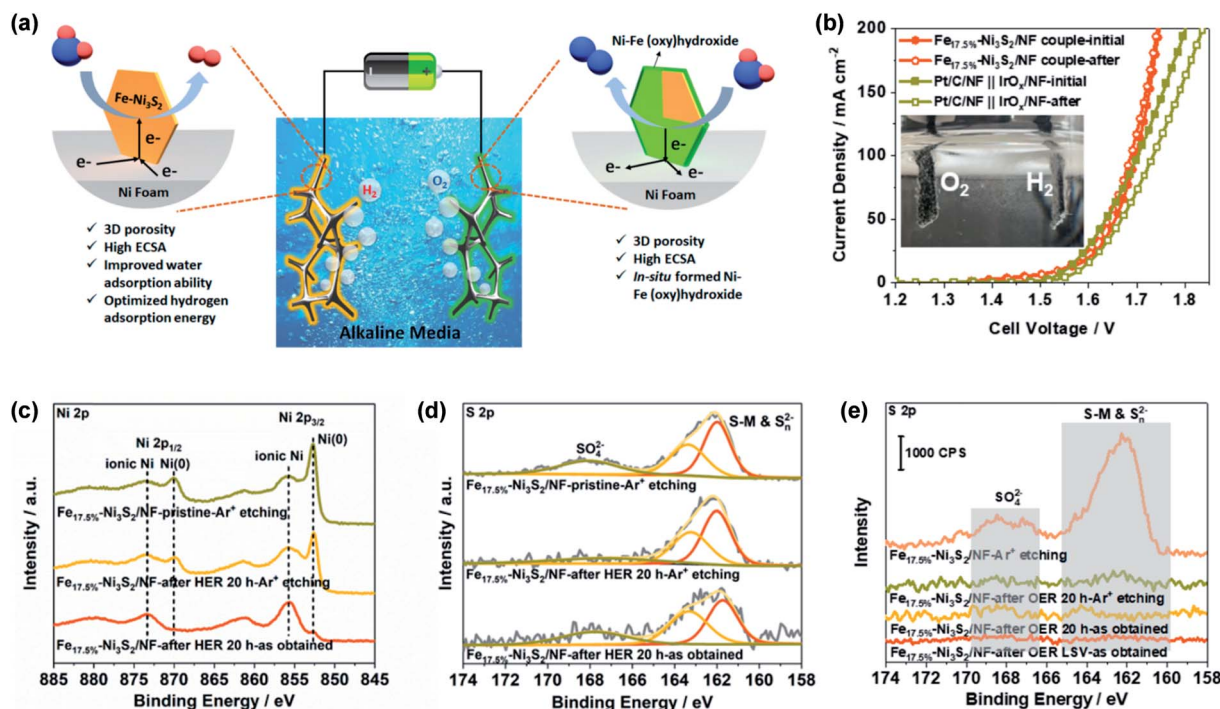


Fig. 5 (a) A schematic diagram of Fe-Ni₃S₂/NF applied to overall water splitting in a two-electrode configuration. (b) The polarization curves of an Fe_{17.5%}-Ni₃S₂/NF couple and Pt/C/NF//IrO_x/NF water electrolyzers in alkaline solution at 5 mV s⁻¹. (c) Ni 2p and (d) S 2p high-resolution XPS spectra of Fe_{17.5%}-Ni₃S₂/NF before and after long-term HER testing. (e) S 2p high-resolution XPS spectra of Fe_{17.5%}-Ni₃S₂/NF before and after long-term OER testing. Ar⁺ etching was employed to remove surface materials.⁸

further mimicked S^{δ-}-TMⁿ⁺ building blocks in chalcogels *via* a simple model based on S^{δ-}-Cⁿ⁺-H₂O clusters to verify the indirect promotion of S sites, where Cⁿ⁺ represents K⁺/Ba²⁺ cations. In this case, S sites were proved to be enhancers of the HER activity of metal sites in alkaline solutions.

The OER is more complex than the HER, involving the formation of many intermediates during the reaction, such as -OH and -OOH intermediates. Subbaraman *et al.* proposed that the increasing OER activities of 3d-M hydr(oxy)oxides (M = Fe, Co, Ni, or Mn) are closely connected with the decreased bond strength of OH_{ad}-Mⁿ⁺, which was mainly ascribed to enhanced repulsion between the metal d-band and coordinated oxygen p-band centers.⁵³ As we know, electronegative S atoms in TMS would hinder the coordination of -OH with metal atoms due to repulsion between the S 3p orbital and O 2p orbital. However, this 3p-2p repulsion could accelerate the further oxidation of -OOH intermediates, and the formation of -OOH intermediates could also be promoted by the delocalized electrons of S atoms.³

There also have been numerous efforts to reveal the true role of the S atoms of TMS in the OER. As a general rule, TMS tend to function as pre-catalysts in the OER because of the conversion of sulfides to transition metal hydr(oxy)oxides during the OER process.⁵⁴ Zhang *et al.* confirmed that Ni-Fe (oxy)hydroxides that formed in the near-surface region of Fe-doped Ni₃S₂ on Ni foam (Fe-Ni₃S₂/NF) are the real active materials that actually contribute to the OER activity.⁸ In this case, the bifunctional Fe-Ni₃S₂/NF electrode could act both as a catalyst for the HER and also as a pre-catalyst for the *in situ* formation of active-

oxidation-state species for electrocatalyzing the OER, which was verified based on high-resolution XPS spectra before and after long-term testing (Fig. 5). No obvious changes appeared in the component types and binding energies of both Ni and S elements, indicating that Fe-Ni₃S₂/NF possesses superior durability in terms of structure and composition during the HER process. Nevertheless, high-resolution S 2p XPS spectra revealed that the signal intensity of S on the surface was obviously reduced after OER testing and even after further Ar⁺ etching, suggesting that near-surface sulfides had been transformed to Ni-Fe (oxy)hydroxides in the OER process.

Coincidentally, Mabayoje *et al.* discovered that NiS can transform into amorphous NiO_x during the OER, which was derived from the finding that S anions in NiS were completely depleted in the active form of the electrocatalyst.⁵⁵ In the Ni 3p XPS region, they speculated that the number of oxidation states of Ni in amorphous NiO_x induced by Ni vacancies are higher than in crystalline NiO. Furthermore, amorphous NiO_x derived from NiS had a relatively high electrochemically active surface area and exhibited enhanced OER activity. Thus, TMS can serve as pre-catalysts to produce more active transition metal hydr(oxy)oxides for electrocatalyzing the OER.

In conclusion, when TMS are employed as electrocatalysts, the main active sites are always metal atoms, while S atoms play pivotal roles in determining the catalytic activity. In the OER process, S atoms can promote the formation of more active transition metal hydr(oxy)oxides. In the HER process, exposed S atoms can act as active sites under specific conditions,

providing places for hydrogen adhesion and separation when metal atoms are the active sites, while S vacancies can induce more active sites and change the coordination environments of adjacent metal atoms, enhancing the activity. A clear understanding of the roles of the S atoms/vacancies of TMS in water splitting could bring enlightenment for designing novel electrocatalysts. Nevertheless, more systematic research revealing the exact roles of these atoms in actual reactions should be conducted, and it could greatly accelerate the development of TMS for water electrolysis.

3. Typical synthetic methods for TMS

The controllable synthesis of TMS with well-defined shapes, sizes, hierarchical structures, and defect states is the initial step in obtaining high-performance bifunctional electrocatalysts. Up to now, various methods have been developed for preparing high-quality TMS, which can be divided into two types, 'bottom-up' and 'top-down' strategies.² A schematic illustration of typical synthetic methods for TMS materials is shown in Fig. 6.

3.1 'Top-down' strategies

Typically, top-down strategies refer to the exfoliation or successive reduction of a bulk starting material to target nanostructures, such as *via* ball milling and sputtering methods. In particular, exfoliation is a simple, low-cost, and promising technique for fabricating high-quality, layered low-dimensional TMS materials; mechanical exfoliation and liquid phase exfoliation are common methods.⁵⁶

Mechanical exfoliation is a clean and easy way to gain pristine TMS of high-quality that are suitable for fundamental studies and potential applications based on the intrinsic thickness-dependent properties of TMS. Typically, bulk TMS

materials with layered structures are applied as starting materials, and TMS parts are peeled off using adhesive tape ('sticky' tape) and then transferred onto the target surface.² The TMS parts attach to the surface *via van der Waals* forces. Li *et al.* produced single-layer and multilayer MoS₂ nanosheets *via* mechanical exfoliation.⁵⁷ However, some challenges still remain in relation to mechanical exfoliation due to the following points: the bulk starting crystals are expensive; a transfer process is essential for further applications; and the as-exfoliated TMS flakes always have random shapes and low yields.

Liquid phase exfoliation is an efficient method for exfoliating TMS in solution; the yields are considerably higher than those obtained *via* mechanical exfoliation, while the product is of lower quality. There are two approaches for exfoliating TMS in the liquid phase. The first one is exfoliation *via* a mechanical method, such as shearing, sonication, stirring, grinding, or bubbling. Exfoliation mechanisms are mainly based on external force induced *via* mechanical effects and interactions with solvent molecules.⁵⁸ Coleman *et al.* reported that layered compounds such as MoS₂ and WS₂ could be easily exfoliated into individual layers *via* sonication exfoliation in solution (Fig. 7a and b).⁵⁹ This procedure was insensitive to water and air and could be potentially scaled up to obtain large quantities of exfoliated products. Additionally, hybrid dispersions or composites could be made for further film-forming *via* blending these materials with suspensions of other nanomaterials or polymer solutions. Later, Varrla *et al.* proposed the large-scale shear-exfoliation of MoS₂ nanosheets in aqueous surfactant solutions using a kitchen blender (Fig. 7c and d).⁶⁰ *Via* optimizing the processing parameters, such as the MoS₂ concentration, mixing time, liquid volume, and rotor speed, high concentrations (0.4 mg mL⁻¹) and high production rates (1.3



Fig. 6 A schematic illustration of typical synthetic methods for TMS materials.



Fig. 7 (a) Photographs of dispersions of MoS₂ and WS₂ in NMP. (b) TEM images of MoS₂ and WS₂ flakes produced *via* sonication exfoliation.⁵⁹ (c) A photograph of a kitchen blender. (d) TEM images of typical MoS₂ flakes produced *via* shearing exfoliation.⁶⁰ (e) A schematic diagram of the preparation of trilayer MoS₂ nanosheets *via* chemical intercalation. (f and g) TEM images of the exfoliated MoS₂ nanosheets produced *via* chemical intercalation.⁶² (h) A schematic illustration of the mechanism of MoS₂-flake formation *via* electrochemical intercalation. (i and j) TEM and HRTEM images of exfoliated MoS₂ flakes produced *via* electrochemical intercalation.⁶³

mg min⁻¹) could be achieved. In addition, the surfactant concentration not only influenced the nanosheet concentration but it also had great effects on the nanosheet lateral size and thickness, with higher surfactant concentrations giving smaller flake sizes. In this regard, the average MoS₂ nanosheet dimensions could be controlled during shear exfoliation, at least in the ranges of about 40–220 nm for length and about 2–12 layers for thickness.

As reported, the surfactant plays a key role in liquid-phase exfoliation, and it could greatly influence the dimensions of the products. Zhao *et al.* prepared WS₂ nanodots with high quality and uniformity that were dispersed in aqueous solution *via* liquid-phase exfoliation from bulk crystals with the aid of ultrasonication in surfactant aqueous solution.⁶⁴ The size distributions of WS₂ nanodots showed that the lateral sizes were 2.7 ± 0.8 nm and the height was 0.7 nm, which was about the size of a single layer. Remarkably, a high concentration of 1T-WS₂ existed in the as-prepared WS₂ nanodots, which could be responsible for its enhanced electrocatalytic activity.

The second liquid-phase exfoliation route involves ionic intercalation, where lithium ions are typically intercalated between the TMS layers to reduce *van der Waals* forces between layers *via* enlarging the interlayer spacing.⁶⁴ Ionic intercalation is beneficial for subsequent exfoliation, and it can involve chemical intercalation and electrochemical intercalation. Fan *et al.* developed an chemical intercalation method *via* the incorporation of sub-stoichiometric amounts of *n*-butyllithium

(BuLi) to controllably exfoliate MoS₂ crystals into trilayer nanosheets (Fig. 7e–g).⁶² The intercalated lithium at the edges of the MoS₂ crystals could serve as a wedge to promote the exfoliation of MoS₂ *via* solvent molecules with a yield of 11–15 wt% exfoliated nanosheets in ethanol/water mixtures. The exfoliation efficiency of a pre-intercalated sample was notably increased by at least 1 order of magnitude compared with the starting MoS₂ microcrystals. However, some shortcomings related to chemical intercalation are inevitable, like extreme sensitivity to ambient conditions and the small lateral size of the chemically exfoliated nanosheets. Liu *et al.* prepared large-area atomically thin MoS₂ nanosheets *via* the electrochemical exfoliation of a bulk MoS₂ crystal (Fig. 7h–j).⁶³ Firstly, ·OH, ·O radicals, and/or SO₄²⁻ anions were inserted into the MoS₂ layers under a positive bias applied at the working electrode, weakening the *van der Waals* interactions between layers. Then, the oxidation of these radicals and/or anions could lead to the release of O₂ and/or SO₂ gas, which could cause the great expansion of the MoS₂ interlayers. Finally, atomically thin MoS₂ flakes were detached from the bulk MoS₂ crystal by the erupting gas and then suspended in solution.

Ball milling is a low-cost and easy scalable mechano-chemical route for preparing TMS nanostructures. In the ball-milling process, powder particles undergo severe mechanical deformation due to collisions with stainless-steel balls and the vessel; they then become continually deformed, cold-welded, and fractured. Concurrently, mechano-chemical reactions



Fig. 8 (a) A schematic diagram of the hydrothermal synthesis of NiS_2 and NiS porous hollow microspheres. (b) An SEM image of NiS_2 hollow microspheres. (c) An SEM image of NiS porous hollow microspheres.⁶⁷ (d–i) FESEM images of CoS synthesized at 160 °C for given reaction times of 2 h, 5 h, 12 h, 24 h, 36 h, and 48 h, respectively.⁶⁸

and/or solid-state reactions can happen in powder blends. The ball-milling time is a key parameter, and an inert gas atmosphere is usually used to protect the TMS powder from oxidation. Homogeneous NiS powders were prepared by ball milling of several mixed phases, such as Ni_3S_2 , Ni_7S_6 , Ni_xS_6 , and Ni_3S_4 , up to 12 h under an Ar atmosphere.⁶⁵ Similarly, Ambrosi *et al.* fabricated a MoS_2 electrocatalyst *via* ball milling bulk natural MoS_2 , and the size reduction contributed to the enhanced electrocatalytic performance.⁶⁶

3.2 'Bottom-up' strategies

Proverbially, bottom-up strategies refer to the build-up of nanomaterials atom by atom, molecule by molecule, or cluster by cluster, for example *via* hydrothermal method, solvothermal method, chemical vapor deposition, electrochemical deposition, atomic layer deposition, thermal deposition, and solid-phase synthesis.

A hydrothermal method is a chemical reaction in water at both high temperature and under high pressure, which usually happened in a sealed pressurized vessel. This method is a simple yet efficacious way to fabricate TMS nanomaterials with controlled compositions and structures, and it can be easily realized *via* adjusting the metal salt precursors and temperature/pressure conditions. Luo *et al.* fabricated NiS_2 hollow microspheres using a hydrothermal approach with $\text{Ni}(\text{NO}_3)_2$ and $\text{Na}_2\text{S}_2\text{O}_3$ solutions as the Ni and S sources, respectively.⁶⁷ In this typical procedure, the precursors were dissolved in water and stirred with ethylene glycol and oxalic acid for 20 min before being transferred into a Teflon-lined

stainless-steel autoclave, then heated at 180 °C for 12 h in an electric oven and cooled to room temperature naturally. Finally, the product was washed with deionized water and ethanol and dried overnight in a vacuum oven at 60 °C. Interestingly, when annealed under a mixed atmosphere, the obtained NiS_2 hollow microspheres could further be transformed into NiS porous hollow microspheres (Fig. 8a–c). Interestingly, Dong *et al.* synthesized hierarchical CoS with flower-like, ball-like, cube-like, and hollowed-out nanostructures *via* the one-pot hydrothermal reaction of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and thiourea.⁶⁸ Thiourea played an important role in the reaction, as it decomposed to produce S^{2-} for the formation of CoS and served as a structure-directing agent to control the growth of CoS crystals. The morphologies of the CoS nanostructures formed could be modified *via* fine-tuning the molar ratios of the solvents and reactants, the reaction temperature, the reaction time, and the ligand type (Fig. 8d–i). The growth process of the flower-like CoS nanostructures could be divided into three stages: the formation of small-sized nanoflakes; the formation of a skeleton sphere *via* the attachment of nanoflakes through an Ostwald ripening mechanism; and the development of hierarchical flower-like nanostructures *via* a dissolution–recrystallization process.

Due to the ease of regulation of the reaction, effective element doping can be achieved easily using the hydrothermal method. Xiong *et al.* prepared Co-doped MoS_2 nanosheets with different Co content levels *via* a facile one-step hydrothermal approach.⁶⁹ During the reaction, $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24} \cdot 4\text{H}_2\text{O}$, thiourea, and $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ were applied as the precursors for the

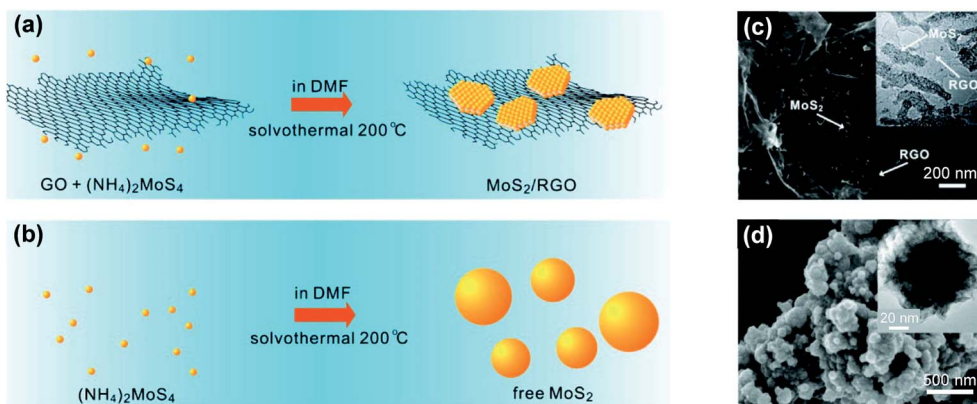


Fig. 9 (a and b) Schematic diagrams of the solvothermal synthesis of MoS_2 in solution with and without graphene sheets. (c and d) SEM and (insets) TEM images of the MoS_2/RGO hybrid and free MoS_2 particles.⁷⁰

hydrothermal preparation of MoS_2 nanosheets covalently doped with Co. The level of Co doping could be regulated simply *via* changing the added amount of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ precursor. Remarkably, a hierarchical flower-like structure with improved conductivity was formed, composed of thin nanosheets, which could be employed as a superior bifunctional electrocatalyst for overall water splitting.

The solvothermal method is very similar to the hydrothermal method, except that an organic solution is applied as the precursor medium instead of water. Li *et al.* prepared MoS_2 nanoparticles on reduced graphene oxide (RGO) sheets *via* the one-step selective solvothermal reaction of $(\text{NH}_4)_2\text{MoS}_4$ and hydrazine in an *N,N*-dimethylformamide (DMF) solution of graphene oxide (GO) sheets at 200 °C (Fig. 9).⁷⁰ During this process, the $(\text{NH}_4)_2\text{MoS}_4$ precursor was reduced to MoS_2 nanoparticles that were laid uniformly and flatly on RGO that was formed from GO upon hydrazine reduction. Importantly, the GO sheets acted as a substrate for the nucleation and subsequent growth of MoS_2 nanoparticles, avoiding MoS_2 coalescing into 3D particles of various sizes. The selective growth of MoS_2 on GO could be attributed to interactions between the Mo precursor and the functional groups of the GO sheets in a suitable solvent environment, because when DMF was replaced with H_2O as the solvent, only two separate phases of MoS_2 and RGO could be obtained. Yuan *et al.* reported the direct growth of Fe-doped Ni_3S_2 nanosheet arrays on a conductive Fe–Ni alloy foil *via* a one-step solvothermal method.⁷¹ A piece of Fe–Ni alloy foil was immersed into a mixture of Na_2S solution and ethanol, and this was then transferred into a Teflon-lined stainless-steel autoclave under N_2 flow. The reaction was carried out at 200 °C for 12 h in an electric oven and the obtained Fe-doped Ni_3S_2 nanosheet arrays grown *in situ* on the surface of the alloy foil could be applied as a self-supported electrode for water oxidation.

As key reaction factors in the solvothermal method, the temperature and precursor concentration exhibit great effects on the structural modification of CoS architectures, as proved by Wang *et al.*⁷² 3D flower-like CoS architectures and CoS microspheres were synthesized through a one-pot solvothermal

method using $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ and thioacetamide as the precursors. According to crystal growth theory, the nucleation and subsequent growth of CoS could be regulated *via* adjusting the supersaturation of the reaction solution, which is related to the precursor concentration and temperature. CoS microspheres could be formed at a high starting precursor concentration, while flower-like CoS architectures were obtained at a low precursor concentration due to limited homogenous nucleation. Likewise, the higher the reaction temperature, the faster the nucleation rate and crystal growth rate. As a result, the controllable fabrication of various CoS nanostructures could be realized *via* a solvothermal route *via* synergistically adjusting the temperature and precursor concentrations, providing a promising way to synthesize TMS with specific morphologies and sizes.

Chemical vapor deposition (CVD) is defined as the deposition of solid materials onto a heated substrate from the vapor phase after several chemical reactions, which is well suited to the preparation of TMS thin films or nanoflakes with high crystallinity on a substrate. During the reaction, solid-state precursors of metal compounds or elemental metal will react with the sulfur source H_2S or vaporized S at high temperatures. Yuan *et al.* reported a facile CVD method to synthesize hexagonal 1T- VS_2 single-crystal nanosheets.⁷³ Typically, a vanadium chloride (VCl_3) precursor in an Al_2O_3 crucible was placed in a quartz tube at the center of the heating zone of a furnace. A SiO_2/Si substrate was placed on top of the crucible to collect the product. Sulfur was placed upstream of the quartz tube and it migrated with the carrier gas, a mixture of N_2 and 10% H_2 , at ambient pressure. The furnace was gradually heated to 550 °C at a heating rate of 20 °C min^{-1} and it was cooled to room temperature naturally after being maintained at high temperature for 10 min. Well-aligned hexagonal and nearly “half-hexagonal” VS_2 nanosheets mostly parallel to the substrate could be obtained *via* tuning the addition of precursors.

Furthermore, in CVD, the reaction pressure and substrate both have great influence on the growth of TMS. Samad *et al.* directly synthesized phase-pure FeS_2 thin films with micron-sized crystalline domains *via* atmospheric pressure CVD with

metallic CoS₂ film as the substrate.⁷⁴ The synthesis of FeS₂ was achieved *via* reacting di-*tert*-butyl disulfide (TBDS) and FeCl₃ precursors at 400–450 °C. CoS₂ has the same cubic crystal structure as FeS₂ with only 2% lattice mismatch between them, so it is highly desirable as a film growth substrate that is unaffected by the sulfur atmosphere and as a conductive substrate for potential electrochemical applications. Subsequently, Shi *et al.* fabricated thickness-tunable TaS₂ flakes and ultrathin centimeter-sized films on gold (Au) foil *via* low-pressure CVD (LPCVD) and atmospheric-pressure CVD (APCVD) routes with TaCl₅ and S as precursors.⁷⁵ Replacing the general SiO₂/Si substrate for TMS growth with Au foil had great advantages, *e.g.*, Au foil could be applied as a metal catalyst for preparing TaS₂ and a conductive current collector of TaS₂-electrode, providing an opportunity to probe either fundamental physical phenomena or applications related to dimensionality effects. Intriguingly, TaS₂ tended to grow into centimeter-sized uniform, several-layer-thick 2H-TaS₂ thin films *via* the LPCVD route, while APCVD was an effective approach to grow TaS₂ with tailored thicknesses due to the excess precursor feeding rate in the synthesis procedure. Schematic illustrations of these two growth processes and the as-prepared TaS₂ are shown in Fig. 10.

For further optimizing the preparation conditions for the CVD growth of TMS, novel methods originating from CVD have been investigated in recent years, like plasma-enhanced CVD (PECVD) and metal-organic CVD (MOCVD). The reaction temperature could be lowered to 150–300 °C using a PECVD technique, thereby making it possible to deposit active TMS on plastic and flexible substrates. Kim *et al.* prepared uniform wafer-scale 1T-WS₂ film *via* a PECVD technique.⁷⁶ Different from common CVD methods, the plasma in an ionized gas state with

ions and radicals does not have high thermal energy. The growth temperature was maintained at 150 °C, enabling the direct preparation of uniform and stable 1T-WS₂ films on both flexible polymer and rigid dielectric substrates. Specifically, organic-metallic precursors are used in MOCVD, and this method can be applied in modern semiconductor industries to prepare single-crystal epitaxial films. Song *et al.* prepared transferable few-layer MoS₂ structures with desired sizes and patterns *via* an MOCVD method with a Au catalyst.⁷⁷ The MOCVD reaction encompassed the following steps: the low-temperature reaction of [Mo(CO)₆] and a pre-deposited Au thin film to form a Mo–Au surface alloy; and the reaction of the Mo–Au surface alloy with H₂S to obtain large-scale MoS₂ atomic layers that could be isolated from the substrate *via* etching the Au films. Importantly, the MoS₂ atomic layers could be fabricated with desired patterns through simply defining the pattern of the Au thin film *via* conventional lithographic techniques.

Solid-phase synthesis has been widely used to convert metal or metal compounds directly to TMS *via* a sulfurization process at high temperatures; it is also a promising method for fabricating self-supported TMS electrodes. Wang *et al.* reported hierarchical WS₂ film that was grown *in situ* on tungsten foil *via* one-step solid-phase synthesis, namely a surface-assisted chemical vapor transport method (Fig. 11a and b).⁷⁸ Tungsten foil and sulfur powder were sealed together in a small quartz ampoule under high vacuum and reacted at high temperature for 10 min to get thin WS₂ film with a hierarchical structure on the surface of the tungsten foil. Tungsten foil acted as the raw material for solid-phase synthesis and as the substrate for the nucleation and growth of WS₂ simultaneously. The strong interaction between WS₂ film and tungsten foil dramatically enhanced the electrical conductivity and electrocatalytic



Fig. 10 (a) A schematic illustration of the LPCVD growth process of TaS₂. (b) An SEM image of 2H-TaS₂ flakes (synthesized at 750 °C for ~10 min) on Au foil. (c) An SEM image of 2H-TaS₂ flakes (synthesized at 750 °C for ~20 min) on Au foil, and an AFM height image of a transferred film edge. (d) A schematic illustration of the APCVD growth process of TaS₂. (e and f) SEM images of 2H-TaS₂ domains on Au foil with growth times of 3 and 5 min. (g) A plot of the tunable thickness of 2H-TaS₂ as a function of the growth time.⁷⁵

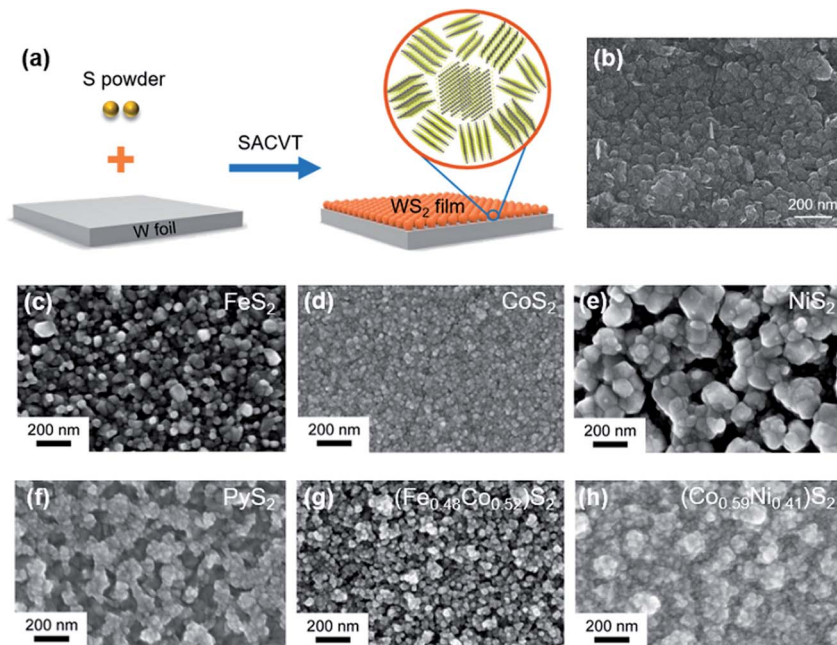


Fig. 11 (a) Schematic diagrams of WS₂ film on tungsten foil prepared via the SACVT method. (b) An SEM image of the WS₂ film.⁷⁸ (c–h) SEM images of FeS₂, CoS₂, NiS₂, PyS₂, (Fe_{0.48}Co_{0.52})S₂, and (Co_{0.59}Ni_{0.41})S₂ thin films on graphite disk substrates prepared via solid-phase synthesis.⁸⁰

durability of this self-supported WS₂ electrode. Jin *et al.* fabricated vertically oriented single-crystal FeS₂ nanowires from steel foil via solid-phase synthesis.⁷⁹ During the sulfurization procedure, low-carbon steel foil, placed near the hot zone of the tube, was heated at 350 °C for 2 h under a sulfur atmosphere that was formed via evaporation from upstream sulfur powder.

What is more, metal sources can be deposited onto other substrates to reduce the utilization of metal. Faber *et al.* prepared pyrite-phase FeS₂, CoS₂, and NiS₂ thin films and thin films of their alloys via a simple solid-phase synthesis reaction (Fig. 11c–h).⁸⁰ Firstly, high-purity metal was deposited on the polished surface of a graphite disk substrate via electron-beam evaporation. To prepare ternary pyrite alloy films, metal bilayers containing two kinds of metals were sequentially evaporated onto graphite disk substrates. Finally, the metalized substrates were loaded into the center of a fused silica tube and heated at 500 °C for 1 h under a sulfur atmosphere until they were converted into their corresponding disulfides. In particular, solid-phase synthesis could also be combined with other methods, such as hydrothermal, solvothermal, and sol-gel methods. Miao *et al.* prepared mesoporous FeS₂ materials with large surface areas using a sol-gel method followed by solid-phase synthesis treatment.⁸¹ Amorphous Fe₂O₃ was first obtained via an inverse micelle sol-gel method, and it was then sulfurized to FeS₂ using H₂S gas and sulfur powder as sulfur sources.

Except for the direct growth of TMS materials on substrates, TMS powder can also be synthesized via solid-phase synthesis. Single-crystal FeS₂ nanorods, nanobelts, and nanoplates have been prepared via the solid-phase synthesis reaction of FeCl₂ or FeBr₂ with sulfur powder at high temperatures.⁸² During solid-phase synthesis reactions, dehydrated FeCl₂ or FeBr₂

precursor powder was heated at 425 °C under a flowing sulfur atmosphere to produce phase-pure cubic iron pyrite. Interestingly, Bara *et al.* investigated the effects of the degree of sulfurization on the preparation of MoS₂ slabs on an α -Al₂O₃ substrate from MoO₃ upon reaction with a H₂S/H₂ gas phase.⁸³ The degree of sulfurization during solid-phase synthesis strongly depended on the surface orientation of the α -Al₂O₃ substrate, and this had a clear impact on the strength of the metal-support interactions in the oxide states with consequences for the sulfurization, stacking, size, and orientation of MoS₂ slabs. Eventually, the maximum degree of sulfurization was about 90% at low temperature (300 °C) for the C (0001) plane with a larger MoS₂ slab size. However, the solid-phase synthesis process in this case was similar to the CVD method.

In summary, numerous effective preparation methods have been established for the fabrication of high-quality TMS materials, which could be divided into ‘bottom-up’ and ‘top-down’ strategies. From the viewpoint of surface morphology, the obtained TMS exhibited different morphologies, *e.g.*, nanodots, nanoparticles, nanosheets, flower/ball/cube-like nanostructures, microspheres, and thin films, which strongly depended on the synthesis method and conditions used. The experimental parameters all play key roles in the controllable synthesis of TMS materials, such as the sulfur/metal source type, the precursor concentration, the solvent type, and the reaction temperature/time. Importantly, self-supported TMS electrodes could be fabricated via the sulfurization of metal substrates. Although various methods have been successfully employed to prepare TMS materials, the growth mechanisms of TMS materials with different morphologies and compositions

remain unclear; these require more investigation in order to deeply understand the underlying behavior.

4. Water splitting properties of TMS

Numerous efforts have been made to develop efficient and inexpensive bifunctional electrocatalysts for electrochemical water splitting. Of these new bifunctional electrocatalysts, TMS have attracted extensive research attention due to their excellent electrocatalytic performance, wide availability, and low cost. Countless TMS have been explored and their great potential for efficiently catalyzing the HER or OER has been revealed. Nevertheless, most of these TMS electrocatalysts can only be utilized for either the HER or OER because of their intrinsic electrochemical properties; this can restrict their electrocatalytic activity and durability, with strong dependence on the pH of the electrolyte. For instance, most HER electrocatalysts are more active in acidic electrolytes, while most effective electrocatalysts for the OER exhibit high solubility in acidic electrolytes and thus are restricted to alkaline media. Consequently, discovering and designing bifunctional electrocatalysts with high activity and durability for both the HER and OER in the same electrolyte is technologically important. This section comprehensively reviewed the intrinsic electrocatalytic performance and optimizing strategies of layered TMS (M = Mo, W, Ta, *etc.*) and non-layered TMS (M = Fe, Co, Ni, *etc.*) for overall water splitting.

4.1 Bifunctional layered TMS (MS₂: Mo, W, Ta, *etc.*) electrocatalysts

4.1.1 Intrinsic electrocatalytic performance. Layered TMS with versatile and tunable properties have been extensively studied as robust substitutes for noble metals for electrocatalyzing the HER. For example, MoS₂,⁸⁴ WS₂,⁷⁸ TaS₂,⁸⁵ and NbS₂ (ref. 86) all can be applied as HER electrocatalysts with excellent activity and stability. Lin *et al.* chemically unzipped WS₂ nanotubes to make metallic WS₂ nanoribbons with increased edge active site content and high electrical conductivity.⁸⁷ The enhanced electrocatalytic HER activity of WS₂ is strongly associated with its well-exposed edge defects. Moreover, Lukowski *et al.* exfoliated metallic 1T-MoS₂ nanosheets *via* lithium intercalation from semiconducting 2H-MoS₂ nanostructures.³⁶ The metallic 1T-MoS₂ nanosheets boosted the electrocatalytic HER activity with a low overpotential of -187 mV at 10 mA cm^{-2} and a small Tafel slope of 43 mV dec^{-1} , showing facile electrode kinetics, low-loss electrical transport, and a proliferated density of active sites. While the basal plane of 2H-MoS₂ is nonreactive, the basal plane of 1T-MoS₂ is active; this mainly arises from the affinity of the surface S sites for binding H species.⁸⁸

Hence, the well-exposed edge sites and good electrical conductivity properties of layered TMS mean they show remarkable intrinsic electrocatalytic HER activities. However, these kinds of TMS are not intrinsically suitable as OER electrocatalysts, and little research has been reported in which they could be directly used for electrocatalyzing the OER. In a rare

example, pure 2H-TaS₂ nanoflakes were reported to have the potential to electrocatalyze both OER and HER processes with overpotentials of 315 mV and 145 mV, respectively, at 10 mA cm^{-2} ,⁸⁹ and MoS₂ microspheres on conductive Ni foam, with enhanced conductivity and accelerated electron transfer, showed great potential for OER applications.⁹⁰ More effort should be devoted to expanding the application of these layered TMS to the OER.

4.1.2 Strategies for optimizing the electrocatalytic performance of layered TMS. To make these layered TMS more suitable for both the HER and OER, numerous efforts have been put forward to optimize their electrochemical properties *via* improving the intrinsic electrical conductivity, enhancing the chemisorption capabilities for O-containing species, and/or improving stability. Practically, three main strategies are used to achieve this goal: phase engineering, composite designing, and heteroatom doping.

Phase engineering is a promising way to regulate the electrocatalytic OER activities of layered multiphase TMS. Traditionally, phase transformation can be realized *via* various strategies, such as chemical exfoliation, ion intercalation, strain engineering, and heteroatom doping. Wu *et al.* exfoliated 2D MoS₂ and TaS₂ nanosheets obtaining both 2H and 1T polymorphs and explored their electrocatalytic OER performances in an acidic medium.⁹¹ The best OER performance, which was comparable to a benchmark IrO₂ electrocatalyst, came from 1T-MoS₂, followed by 1T-TaS₂, 2H-MoS₂, and 2H-TaS₂, unraveling the polymorphic dependence of the electrocatalytic OER activity. A theoretical study also revealed the same electrocatalytic activity trend, in which the 1T polymorph showed higher activity than the 2H layered TMS counterpart. Additionally, Wei *et al.* found that the spontaneous phase transformation of MoS₂ from the 2H to 1T phase was triggered by strong metal-support interactions during iridium (Ir) adsorption on the surface of 2H-MoS₂.⁹² Strong charge transfer between MoS₂ and the adsorbed Ir atoms could promote the phase transformation of MoS₂ and enhance the electrocatalytic activity for overall water splitting, which was superior to even commercial Pt/C and IrO₂ electrodes. The significantly improved electrocatalytic performance could be ascribed to enhanced electrical conductivity and hydrophilicity, increased numbers of interfaces and active sites, and the activated basal plane of MoS₂. The strong synergistic and coupling effects between 1T-MoS₂ and Ir atoms markedly accelerated the reaction kinetics in alkaline solution.

Composite designing is the most attractive strategy for obtaining layered TMS-based bifunctional electrocatalysts for overall water splitting, and it involves integrating layered TMS with other active materials to construct novel heterostructures or composites. As we know, layered TMS can be employed as excellent HER electrocatalysts while they exhibit poor electrocatalytic OER performance. In turn, first-row metal compounds generally are preferred alternatives to noble metal oxides (IrO₂, RhO₂, or RuO₂) in the OER, such as metal sulfides, oxides, and hydroxides.²⁴ Hence, assembling a layered TMS with high activity for the HER with an active OER electrocatalyst in one system is a good choice for building highly effective

bifunctional electrocatalysts. The newly formed composite/heterostructure would possess enhanced electrochemical water-splitting performance due to the synergistic effect of the counterparts.

Based on this design concept, Zhang *et al.* fabricated MoS₂/Ni₃S₂ heterostructures with abundant interfaces on nickel (Ni) foam *via* decorating MoS₂ nanosheets on the surfaces of Ni₃S₂ nanoparticles (Fig. 12a–c).⁹³ The novel MoS₂/Ni₃S₂ heterostructures possessed both highly efficient OER and HER activities in alkaline solution, with overpotentials at 10 mA cm⁻² as low as 218 mV and 110 mV, respectively. Furthermore, an alkaline electrolyzer assembled using the as-prepared MoS₂/Ni₃S₂ heterostructures could deliver a current density of 10 mA cm⁻² at a low cell voltage of 1.56 V. Based on DFT calculations, the constructed interfaces between MoS₂ and Ni₃S₂ could synergistically facilitate the chemisorption of hydrogen- and oxygen-containing intermediates, thus enhancing the electrocatalytic performance for water splitting. Subsequently, Peng *et al.* prepared three-dimensional (3D) Co₉S₈/WS₂ array films on titanium (Ti) foil, in which WS₂ nanosheets were uniformly grown *in situ* on the surfaces of Co₉S₈ array structures to form hybrids with large electrochemically active surface areas.⁹⁴ Co₉S₈/WS₂ array films displayed superior electrocatalytic activity for both the HER and OER with a low cell voltage of 1.65 V at 10 mA cm⁻² when acting as both the cathode and the anode. The good performance for water splitting was ascribed to the unique porous structure and the synergistic effect of electronic and chemical coupling between active WS₂ and Co₉S₈, which resulted in suitable kinetics and energetics for hydrogen adsorption and H₂ formation processes and facilitated electron transport during water electrolysis. To further strengthen the activity and durability, the construction of oxygenated-sulfide hetero-nanosheets was subsequently put

forward. Hou *et al.* grew vertically aligned oxygenated-MS₂-MoS₂ (M = Fe, Co, or Ni) hetero-nanosheets on carbon fiber cloth using bimetal precursors through a hydrothermal strategy.⁹⁵ Among these samples, O-CoS₂-MoS₂ hetero-nanosheets presented the best performance, with low overpotentials of 97 and 272 mV to deliver a current density of 10 mA cm⁻² for the HER and OER, respectively, in alkaline electrolyte. This unique architecture with well-exposed active hetero-interfaces, ultra-fast charge-transport channels, and abundant active sites was highly favorable for obtaining enhanced electrocatalytic performance, which was derived from the synergistic effect of heterostructures. When assembled in a two-electrode electrolyzer, O-CoS₂-MoS₂ hetero-nanosheets merely needed a low cell voltage of 1.6 V to reach a current density of 10 mA cm⁻².

Motivated by this research progress relating to active bimetal transition-metal sulfides, Yang *et al.* proposed the construction of 3D composites co-assembled from MoS₂ and Co₉S₈ nanosheets that were attached to Ni₃S₂ nanorod arrays.⁹⁷ The as-prepared MoS₂/Co₉S₈/Ni₃S₂ supported by Ni foam showed controllable morphologies and compositions, and it demonstrated excellent bifunctional electrocatalytic performance for both the HER and OER over a full pH range. Notably, the optimal sample could reach 10 mA cm⁻² at very low overpotentials of 103, 117, and 113 mV for the HER in alkaline, neutral, and acidic solutions, respectively, while it required extremely low overpotentials of 166, 228, and 405 mV for the OER. In addition, a two-electrode electrolyzer assembled from MoS₂/Co₉S₈/Ni₃S₂ composites also only required low cell voltages of 1.54, 1.80, and 1.45 V to obtain a current density of 10 mA cm⁻² in alkaline, neutral, and acidic media, respectively. The superior performance of the all-pH bifunctional MoS₂/Co₉S₈/Ni₃S₂ electrocatalyst was attributed to enhanced electron

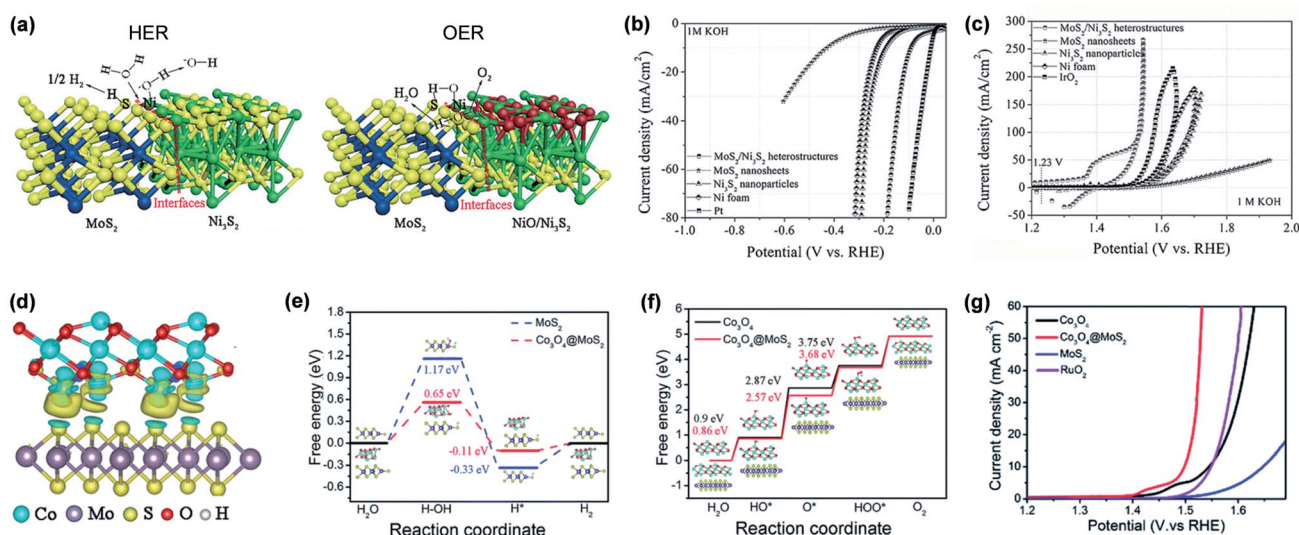


Fig. 12 (a) The proposed HER and OER mechanisms on MoS₂/Ni₃S₂ heterostructures; S: yellow, Mo: blue, Ni: green, H: white, and O: red. (b) Polarization curves and (c) CV curves from MoS₂/Ni₃S₂ heterostructures, MoS₂ nanosheets, Ni₃S₂ nanoparticles, Ni foam, and IrO₂/Pt.⁹³ (d) A schematic diagram of interfacial electron transfer between MoS₂ and Co₃O₄. (e) A free energy diagram of the HER on Co₃O₄@MoS₂ and MoS₂. (f) A free energy diagram of the OER on Co₃O₄@MoS₂ and Co₃O₄. (g) Polarization curves of MoS₂, Co₃O₄, and Co₃O₄@MoS₂ in 1 M KOH.⁹⁶

and mass transfer, and the well-exposed active sites of MoS₂ and Co₉S₈ supported by a substrate with a 3D frame. The development of universal-pH bifunctional electrocatalysts has great significance for industrial applications involving seawater electrolysis.

As well as metal sulfides, layered TMS can also be integrated with other synergistic materials into well-designed architectures to achieve highly effective bifunctional electrocatalysts, like metal oxides, metal carbonitrides, and metal oxyhydroxides. Wang *et al.* innovatively prepared a WS₂ nanosheet array on carbon cloth and then decorated NiO@Ni onto the surface *via* electrodeposition and subsequent thermal oxidation.⁹⁸ This NiO@Ni/WS₂ synergistic electrocatalyst possessed Pt-like activity for the HER with only a low overpotential (40 mV) being needed to drive 10 mA cm⁻². Moreover, when serving as an anode for the OER, it required just 347 mV to deliver a current density of 50 mA cm⁻², which was superior to even commercial RuO₂. Excitingly, an electrolyzer with a two-electrode setup based on NiO@Ni/WS₂ composites could afford 10 mA cm⁻² at an extremely low cell voltage of 1.42 V. The marvelous enhanced bifunctional electrocatalytic performance of the NiO@Ni/WS₂ composite may arise from the unique nanosheet array structure and multicomponent synergetic effects among WS₂, NiO, and Ni. What is more, Liu *et al.* designed Co₃O₄@MoS₂ heterostructures with balanced HER and OER performances.⁹⁶ Theoretical and experimental results revealed that the heterostructures synergistically favored reducing the energy barrier for initial water dissociation, optimizing subsequent H adsorption/desorption for the HER, and enhancing the adsorption of oxygen intermediates for the OER (Fig. 12d–g). Analogously, Ji *et al.* grafted thin MoS₂ nanosheets onto MOF-derived porous Co–N–C flakes and grew them on carbon nanofibers to get excellent bifunctional CoNC@MoS₂/CNF films.⁹⁹ The resultant CoNC@MoS₂/CNFs with hierarchical structures and superior flexibility exhibited remarkable activities and stabilities for both the HER and OER in 1 M KOH. Regarding water electrolysis, a low cell voltage of 1.62 V was required to generate a current density of 10 mA cm⁻², which could be ascribed to the high activity and rapid mass transport derived from a synergistic effect between MoS₂ and Co–N–C flakes. Undoubtedly, MoS₂ as a remarkable and representative layered TMS should be widely investigated to exploit its capacity for water splitting.

The stable phase of MoS₂ is the thermally preferred 2H phase, while free-standing 1T-MoS₂ suffers from an unstable nature. Integrating MoS₂ with other active materials can not only improve the electrocatalytic activity significantly but can also stabilize unstable 1T-MoS₂ *via* serving as a substrate. Shang *et al.* embedded atomically thin 1T-MoS₂ in quasi-amorphous CoOOH *via* a facile one-pot method to achieve a bifunctional electrocatalyst for overall water splitting.¹⁰⁰ The synthesized 1T MoS₂–CoOOH heterostructure exhibited highly efficient activity, with overpotentials of 158 mV in 0.5 M H₂SO₄ for the HER and 345 mV in 1 M KOH for the OER at 10 mA cm⁻². In this heterostructure, the vertical 1T-MoS₂ monolayer contributed mainly to the excellent HER activity, while the CoOOH nanosheets could provide superior OER activity. Importantly, the

CoOOH nanosheets could also anchor and stabilize the 1T-MoS₂ monolayer, achieving a stable architecture, *via* acting as an electron donor to facilitate the phase transition and stabilization of the metallic phase, which also paved the way to the controllable synthesis of thermodynamically unstable phases of layered TMS.

Heteroatom doping is an effective strategy for regulating the fundamental properties of layered TMS and constructing efficient bifunctional electrocatalysts. On one hand, the heteroatoms can directly break the periodic crystalline structure of materials *via* generating defects and distortions in the lattices of layered TMS; this is mainly due to differences between the atomic radii of host atoms and heteroatoms. These defects and distortions can create more active sites for electrocatalytic reactions and increase the reactivity of the surface.¹⁰¹ On the other hand, the introduction of heteroatoms can modulate the surface electronic structure of layered TMS through breaking the original charge balance, building a new charge distribution, and generating localized charge aggregation. The modulated electronic structure could further optimize the chemical adsorption energies of reactive intermediates and change the reaction pathways during water splitting.¹⁰² All these advantageous effects can promote the formation of electrocatalysts with bifunctionality and enhanced performance for water splitting. Heteroatom doping strategies can be divided mainly into two types: substitutional doping and surface charge transfer doping.¹⁰³

In the case of substitutional doping, metal atoms or S atoms in layered structures are substituted with heteroatoms with different valence electrons. Xiong *et al.* developed a covalent Co doping strategy to induce bifunctionality into MoS₂ for effective water splitting (Fig. 13c).¹⁰⁴ Co atoms replaced Mo atoms to obtain MoS₂ covalently doped with Co, which possesses enhanced intrinsic conductivity, a decreased ΔG_{H} value upon optimizing the electronic structure of MoS₂, and improved OER activity, which was ascribed to Co species in a high valence state under anodic potentials. Consequently, the best-performing MoS₂ covalently doped with Co exhibited overpotentials of 48 and 260 mV to afford a current density of 10 mA cm⁻² during the HER and OER, respectively, in 1.0 M KOH. A resultant electrolyzer assembled from MoS₂ covalently doped with Co could generate 4.5 and 9.1 $\mu\text{mol min}^{-1}$ O₂ and H₂, respectively, with nearly 100% faradaic efficiency. Specially, it should also be noted that the covalent doping of Ni and P into 1T-enriched MoS₂ could significantly activate the basal planes, expand the interlayer spacing, and increase the edge density of 1T-enriched MoS₂ in bifunctional nanostructures for water splitting.¹⁰⁵ Further, Xue *et al.* prepared bifunctional Fe-doped MoS₂ nanocanopies on Ni foam (Fe-MoS₂/NF) *via* an *in situ* solvothermal reaction.¹⁰⁶ Fe doping would modify the electronic structure of Mo, and some Fe would likely be present as iron sulfide. Strong coupling interactions between Fe and MoS₂ markedly promote the electrocatalytic activity. The Fe-MoS₂/NF electrode exhibited an OER overpotential of 230 mV at 20 mA cm⁻² and a HER overpotential of 153 mV at 10 mA cm⁻² in 1.0 M KOH.

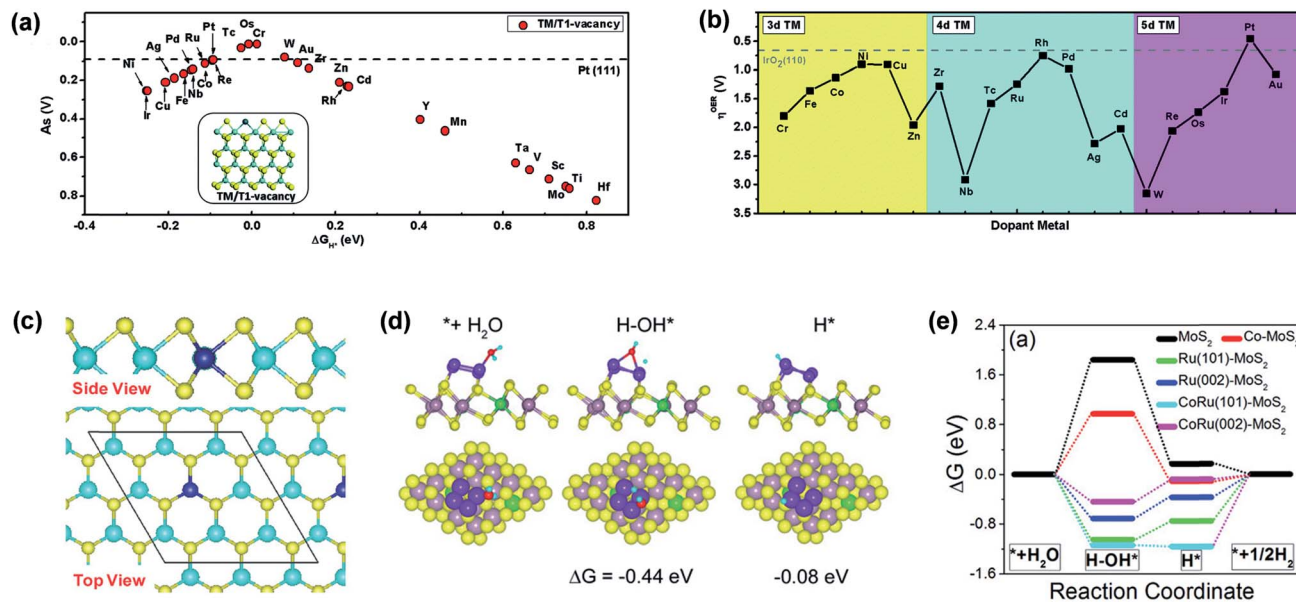


Fig. 13 (a) HER activity as a function of the Gibbs free adsorption energy of hydrogen and the configuration of TM/T1-vacancies. (b) The OER overpotentials of TM-doped T1 vacancy terminations; TM is a transitional metal and a T1 vacancy means that 37.5% of S atoms are added to initial Mo termini.¹⁰⁷ (c) Side and top views of the optimized structure of MoS₂ covalently doped with Co; Mo: green, S: yellow, and Co: blue.¹⁰⁴ (d) The optimized geometries of CoRu(002)–MoS₂ with absorbed intermediates; Mo: grayish purple, S: yellow, Ru: purple, Co: green, H: cyan, and O: red. (e) Gibbs free energy diagrams of the H₂ production pathway (via the Volmer–Heyrovsky reaction in alkaline solution) on different samples.¹⁰⁸

Surface charge transfer doping is realized *via* electron exchange between layered TMS and materials adsorbed on their surfaces. The introduction of doped heteroatoms is a viable method for triggering the bifunctional electrocatalytic activity of layered TMS, as has been proved *via* theoretical methods. Cheng *et al.* screened 28 transition-metal single-atom catalysts (SACs) supported on the edges of MoS₂ as bifunctional electrocatalysts based on DFT calculations (Fig. 13a and b).¹⁰⁷ These SACs could trigger OER activity and maintain the HER performance of MoS₂ edges. The OER activity was calculated *via* a simple equation including the electronegativity of the atoms in the local structure and the coordination number of active metal centers. The lowest theoretical overpotentials for the HER and OER were derived from MoS₂ edges modified with Pt single atoms. Pt single atoms enhanced the intrinsic HER activity of MoS₂ *via* modifying the bonds between S/Mo atoms and H and improved the OER activity *via* influencing the coordination environment and adsorption energy. This work provided theoretical guidance for developing innovative bifunctional electrocatalysts for overall water splitting.

Additionally, the synergistic effect of substitutional doping and surface charge transfer doping could lead to the great enhancement of the electrocatalytic performance of layered TMS. Kwon *et al.* doped metallic 1T'-MoS₂ with Co atoms and further decorated this material with Ru nanoparticles (size = 1–5 nm) to prepare unique Co–Ru–MoS₂ nanosheets for overall water splitting (Fig. 13d and e).¹⁰⁸ Co atoms substituted for Mo atoms homogeneously and Ru nanoparticles were deposited on the surface of Co-doped MoS₂. The enhanced HER performance in 1.0 M KOH was characterized based on the low overpotential

of 52 mV at 10 mA cm⁻² and small Tafel slope of 55 mV dec⁻¹, and this was attributed to favorable water adsorption and dissociation on Ru atoms and advantageous reaction intermediates being catalyzed by Co atoms. The high OER performance benefitted from active RuO₂ that was generated due to the oxidation of Ru nanoparticles, exhibiting a low overpotential of 308 mV at 10 mA cm⁻² and a small Tafel slope of 50 mV dec⁻¹. In short, the synergistic effect of doped Co atoms and active Ru nanoparticles could bring about excellent bifunctional electrocatalytic performance for overall water splitting *via* producing optimized electronic structures.

In conclusion, layered TMS exhibit excellent intrinsic electrocatalytic HER activities due to their well-exposed edge sites and good electrical conductivity, while their bifunctional electrocatalytic performances for water electrolysis could be optimized *via* some strategies such as phase engineering, composite designing, and heteroatom doping. Currently, MoS₂-based and WS₂-based bifunctional electrocatalysts are the most widely investigated materials in the layered TMS family, and they were always coupled with other active materials. To reach a current density of 10 mA cm⁻² in the HER, a NiO@Ni/WS₂ composite only requires an overpotential of 40 mV with a low Tafel slope of 83.1 mV dec⁻¹ in 1.0 M KOH, which is even superior to Pt.⁹⁸ To obtain 10 mA cm⁻² in the OER, the best performance is shown by a MoS₂/Co₉S₈/Ni₃S₂ composite, which only needs a small overpotential of 116 mV in 1.0 M KOH.⁹⁷ A summary of current layered TMS-based bifunctional electrocatalysts is shown in Table 1, and further investigations should put more effort into the design of novel composites.

Table 1 A summary of layered TMS-based bifunctional electrocatalysts for overall water splitting

| TMS-based material | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential η (mV) | Tafel slope (mV dec ⁻¹) | Cell voltage (V @ mA cm ⁻²) | Stability |
|---|---|--|-----------------------------------|---------------------------|-------------------------------------|---|-------------|
| 2H-TaS ₂ nanoflakes ⁸⁹ | Solid-state + hydrothermal | 1 M KOH | HER/10, OER/10 | 145, 315 | 121, 81 | — | — |
| Co ₉ S ₈ /WS ₂ array films ⁹⁴ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 138, — | 80.2, — | 1.65 | 24 h |
| NiO@Ni/WS ₂ (ref. 98) | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/50 | 40, 347 | 83.1, 108.9 | 1.42 | 40 h |
| MoS ₂ /Ni ₃ S ₂ heterostructures ⁹³ | Solvothermal | 1 M KOH | HER/10, OER/10 | 110, 218 | 83, 88 | 1.56 | 10 h |
| MoS ₂ /Ni ₃ S ₂ heteronanorods ¹⁶¹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 98, 248 | 61, 57 | 1.50 | 48 h |
| MoS ₂ /Ni ₃ S ₂ nanorod arrays ¹⁴⁴ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 187, 217 | 90, 38 | 1.467 | 24 h |
| MoS ₂ /Ni ₃ S ₂ /Ni foam ¹⁶² | Electrodeposition + solvothermal | 1 M KOH | HER/10, OER/10 | 99, 185 | 71, 46 | 1.50 | 48 h |
| MoS ₂ /NiS ₂ nanosheets ¹⁵¹ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 62, 278 | 50.1, 91.7 | 1.59 | 24 h |
| MoS ₂ /NiS yolk-shell microspheres ¹⁶³ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 244, 350 | 97, 108 | 1.64 | 24 h |
| Ni ₃ S ₂ -MoS _x /Ni foam ¹⁶⁴ | Solvothermal | 1 M KOH | HER/10, OER/50 | 65, 312 | 81, 103 | 1.54 | 20 h |
| CoNC@MoS ₂ /CNF films ⁹⁹ | Carbonization + solvothermal | 1 M KOH | HER/10, OER/10 | 143, 350 | 68, 51.9 | 1.62 | 56 h |
| MoS ₂ /Co ₉ S ₈ /Ni ₃ S ₂ composites ⁹⁷ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 113, 116 | 85, 58 | 1.54 | 24 h |
| | | 0.5 M H ₂ SO ₄ | HER/10, OER/10 | 103, 228 | 55, 78 | 1.45 | 80 min |
| | | 1.0 M PBS | HER/10, OER/10 | 117, 405 | 56, 71 | 1.80 | 20 h |
| MoS ₂ /Co ₉ S ₈ /N-carbon heterostructures ¹⁶⁵ | Solvothermal | 1 M KOH | HER/10, OER/10 | 95, 230 | 94, 77 | 1.63 | 25 h |
| 1T MoS ₂ -CoOOH ¹⁰⁰ | Hydrothermal | 0.5 M H ₂ SO ₄ , 1 M KOH | HER/10, OER/10 | 158, 345 | 42, 59 | — | 20 h |
| O-CoS ₂ -MoS ₂ heteronanosheets ⁹⁵ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 97, 272 | 70, 45 | 1.60 | 10 h |
| CoS _x @MoS ₂ heterostructures ¹⁶⁶ | Solvothermal | 1 M KOH | HER/10, OER/10 | 146, 276 | 56.5, 26.1 | 1.668 | 20 h |
| Ni-Fe-LDH/MoS ₂ superlattices ¹⁶⁷ | Exfoliation + assembly | 1 M KOH | HER/10, OER/10 | 180, 250 | 67, 45 | 1.4 | 11 h |
| (Ni, Fe)S ₂ @MoS ₂ heterostructures ¹⁶⁸ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 130, 270 | 101.22, 43.21 | 1.56 | 45 h |
| 1T'-MoS ₂ /(Co, Fe, Ni) ₉ S ₈ nanotube arrays ¹¹⁸ | Solvothermal | 1 M KOH | HER/10, OER/10 | 58, 184 | 37.5, 49.9 | 1.429 | 80 h |
| MoS ₂ -NiS ₂ /N-doped graphene ¹⁶⁹ | CVD | 1 M KOH | HER/10, OER/10 | 172, 370 | 70, — | 1.64 | 24 h |
| CoS-Co(OH) ₂ @MoS _{2+x} (ref. 170) | Hybridization | 1 M KOH | HER/10, OER/10 | 143, 380 | 68, 68 | 1.58 | 11 h |
| CoS ₂ -C@MoS ₂ nanofibers ¹⁷¹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 173, 391 | 61, 46 | — | 1000 cycles |
| CoS ₂ -MoS ₂ hollow spheres ¹⁷² | Solvothermal | 1 M KOH | HER/10, OER/10 | 109, 288 | 52, 62.1 | 1.61 | 10 h |
| Co ₃ S ₄ @MoS ₂ heterostructures ¹⁷³ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 136, 280 | 43, 74 | 1.58 | 10 h |
| Co ₉ S ₈ @MoS ₂ /carbon nanofibers ¹²⁰ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 190, 430 | 110, 61 | — | 12 h |
| Co ₃ O ₄ /MoS ₂ heterostructures ⁹⁶ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 90, 269 | 59.5, 58 | 1.59 | 12 h |
| Co ₃ O ₄ /MoS ₂ heterostructures ¹⁷⁴ | Hydrothermal | 1 M KOH | HER/10, OER/20 | 205, 230 | 98, 45 | — | 14 h |
| Ni ₂ P/MoO ₂ @MoS ₂ nanomaterials ¹⁷⁵ | Solid-phase synthesis + phosphorization | 1 M KOH | HER/10, OER/10 | 159, 280 | 77, 85 | 1.72 | 40 h |

Table 1 (Contd.)

| TMS-based material | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential η (mV) | Cell voltage | | Stability |
|---|---|--------------------------------------|--------------------------------------|------------------------------|--|------------------------|-----------|
| | | | | | Tafel slope (V @ (mV dec ⁻¹)) | (mA cm ⁻²) | |
| Phosphorene quantum dot/ MoS ₂ nanosheets ¹⁷⁶ | Electrochemical | 0.1 M KOH | HER/10, OER/10 | 600, 370 | 162, 46 | — | 15 h |
| N-Ni ₃ S ₂ /N-MoS ₂ heteronanowires ¹⁷⁷ | Solid-phase synthesis | 1 M KOH | HER/10, OER/200 | 54, 387 | 104.5, 101.2 | 1.79 @ 100 | 15 h |
| MoS ₂ /NiS nanosheets ¹⁷⁸ | Solvothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/15 | 92, 271 | 113, 53 | 1.61 | 80 h |
| Co ₃ S ₄ @MoS ₂ -Ni ₃ S ₂ nanorods ¹⁷⁹ | Hydrothermal | 1 M KOH | HER/10, OER/50 | 136, 270 | 72, 69 | — | 30 h |
| MoS ₂ /NiCoS nanosheets ¹⁸⁰ | Solvothermal | 1 M KOH | HER/10, OER/10 | 189, 290 | 75, 77 | 1.5 | 22 h |
| MoS ₂ /Fe ₃ Ni ₄ S ₈ heterostructures ¹⁸¹ | CVD | 1 M KOH | HER/10, OER/10 | 120, 204 | 45.1, 28.1 | — | 10 h |
| Amorphous Ni-Co complexes/ 1T-MoS ₂ (ref. 182) | Solvothermal | 1 M KOH | HER/10, OER/10 | 70, 235 | 38.1, 45.7 | 1.44 | 48 |
| MoS ₂ /(Co, Ni and Fe) sulfide ¹⁸³ | Solvothermal | 1 M KOH | HER/100, OER/10 | 359, 178 | 76, 35.7 | — | 32 h |
| Fe-MoS ₂ /Ni ₃ S ₂ (ref. 184) | Solvothermal | 1 M KOH | HER/10, OER/10 | 130.6, 256 | 112.7, 59.5 | 1.61 | 180 h |
| Co-MoS ₂ (ref. 104) | Hydrothermal | 1 M KOH | HER/10, OER/10 | 48, 260 | 52, 85 | — | 11 h |
| Fe-MoS ₂ nanocanopies ¹⁰⁶ | Solvothermal | 1 M KOH | HER/10, OER/20 | 153, 230 | 85.6, 78.7 | 1.52 | 140 h |
| Co-Ru-MoS ₂ nanosheets ¹⁰⁸ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 52, 308 | 55, 50 | 1.67 @ 20 | 16 h |
| Ir/MoS ₂ nanosheets ⁹² | Hydrothermal | 1 M KOH | HER/10, OER/10 | 44, 330 | 32, 44 | 1.57 | 12 h |
| Co-MoS ₂ nanosheets ⁶⁹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 90, 190 | 50.3, 64.7 | 1.58 | — |
| | | 0.5 M H ₂ SO ₄ | HER/10, OER/10 | 60, 540 | 64.7, 245.9 | 1.90 | ~11 h |

4.2 Bifunctional non-layered TMS (M_xS_y: Fe, Co, Ni, etc.) electrocatalysts

Non-layered TMS electrocatalysts are the most extensively studied bifunctional electrocatalysts for overall water splitting and they include FeS₂, Ni₃S₂, and CoS₂. Composed of first-row transition metals and S elements, these kinds of materials are very widely available, inexpensive, and chemically stable, making them quite valuable for energy conversion applications.⁸⁰ Such bifunctional electrocatalysts based on Fe, Co, and Ni have been widely reported, frequently with similar or, in some cases, even better activities for both the HER and OER compared with electrocatalysts based on noble metals (like Pt, Ir, and Ru). To further improve their electrocatalytic performances for overall water splitting, many strategies have been developed, such as synergetic composite designing, heteroatom doping, strain engineering, facet engineering, edge engineering, and defect engineering.

4.2.1 Intrinsic bifunctional electrocatalysts. Ni-based sulfides are highly efficient non-noble metal-based bifunctional electrocatalysts with high conductivity and unique configurations, including Ni₃S₂, Ni₃S₄, and NiS.¹⁰⁹ The controllable preparation of NiS_x compounds with different phases and compositions can be easily realized *via* adjusting the added

precursor ratios. Shi *et al.* prepared three NiS_x compounds with large surface areas *via* a hydrothermal method, namely hexagonal NiS and cubic Ni₃S₄ and NiS₂.¹¹⁰ Upon increasing the S/Ni molar ratio in the final product, the particle size decreases and the surface of the NiS_x compound becomes rougher. A pyrite-type NiS₂ electrocatalyst with enriched Ni³⁺ sites on the surface exhibited the best electrocatalytic activity, which was characterized by low overpotentials of 241 and 147 mV at 10 mA cm⁻² for the OER and HER, respectively, in 1.0 M KOH. In practice, the enhanced performance could be attributed to the high-valence-state Ni³⁺ sites, which could optimize the binding energies with H* and OH* species, reduce the reaction barriers, and consequently speed up the sluggish kinetics. Similarly, Zheng *et al.* synthesized a series of NiS_x nanocrystals with controllable phases and compositions *via* a facile polyol solution process, namely NiS, Ni₃S₂, and NiS₂.¹¹¹ Ni₃S₂ exhibited superior OER and HER activities compared to NiS and NiS₂ in alkaline media, requiring overpotentials of 295 mV and 112 mV to afford 10 mA cm⁻² for the OER and HER, respectively. The higher proportion of Ni in Ni₃S₂ leads to intrinsic metallic conductivity, abundant active sites, and optimized ΔG_{H} values for the HER, additionally generating more active NiOOH species on the surface of Ni₃S₂ for the OER.

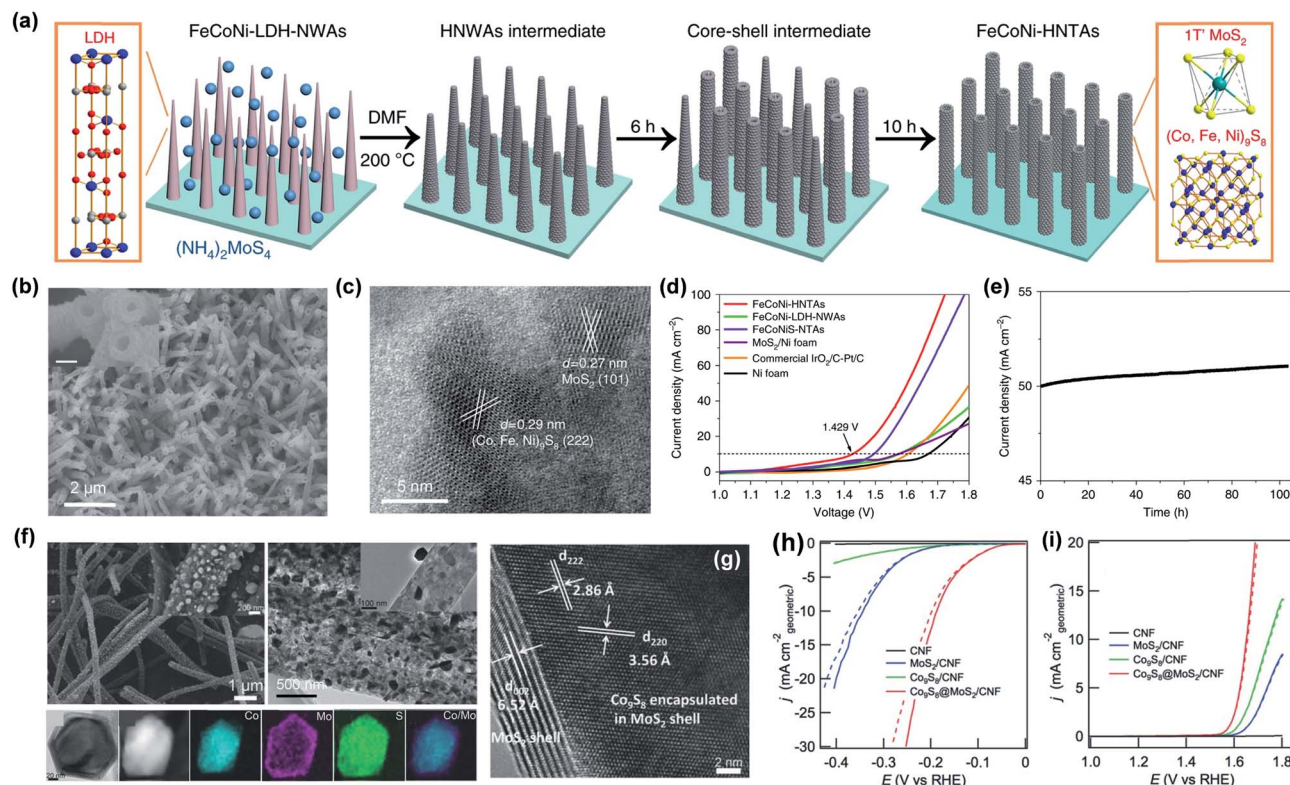


Fig. 14 (a) Schematic diagrams of the preparation process of FeCoNi-HNTAs. (b) An FESEM image and (c) HRTEM image of the as-prepared FeCoNi-HNTAs. (d) Polarization curves and (e) the chronoamperometric response of FeCoNi-HNTAs in 1 M KOH with a typical two-electrode set-up.¹¹⁸ (f) FESEM, TEM, STEM, and STEM-EDS element mapping images of a $\text{Co}_9\text{S}_8@\text{MoS}_2$ nanocrystal. (g) A HRTEM image of the $\text{Co}_9\text{S}_8@\text{MoS}_2$ nanocrystal. Polarization curves of $\text{Co}_9\text{S}_8@\text{MoS}_2/\text{CNFs}$, MoS_2/CNFs , $\text{Co}_9\text{S}_8/\text{CNFs}$, and CNFs for (h) the HER in 0.5 M H_2SO_4 and (i) the OER in 1 M KOH.¹²⁰

Obviously, Ni_3S_2 shows superior electrocatalytic performance owing to the high Ni element content, which could bring about intrinsic metallic behavior and a continuous network of Ni–Ni bonds throughout its whole structure. Zeng *et al.* reported 3D hierarchical Ni_3S_2 superstructures grown *in situ* on Ni foam *via* a chemical etching method.¹¹² The porous rod-like array structure of Ni_3S_2 could provide large active surface areas, porosity, and highly effective accessibility. This desirable self-supported electrode offered $\sim 100\%$ Faradaic yields for both the HER and OER, and it exhibited remarkable electrocatalytic stability for more than 50 h in 1.0 M KOH. Coincidentally, Zhang *et al.* fabricated 3D coral-like Ni_3S_2 on Ni foam ($\text{Ni}_3\text{S}_2/\text{NF}$) *via* a one-step electrochemical method.¹¹³ The optimal $\text{Ni}_3\text{S}_2/\text{NF}$ electrode exhibited high electrocatalytic activity for overall water splitting, with low overpotentials of 89 and 242 mV for the HER and OER, respectively, to afford 10 mA cm^{-2} . When the as-prepared $\text{Ni}_3\text{S}_2/\text{NF}$ electrodes were assembled in an electrolyzer, a low cell voltage of 1.577 V was required to get 10 mA cm^{-2} , with extremely long-term durability.

Ni-based sulfides can use commercial Ni foam directly as both a raw material and a growth template; the *in situ* growth of active Ni-based sulfides on Ni foam could further result in a self-supporting electrode for overall water splitting. Zhu *et al.* reported NiS microsphere film grown *in situ* on Ni foam (NiS/NF) *via* a vapor sulfurization reaction.¹¹⁴ This NiS/NF electrode

could deliver 20 mA cm^{-2} with an overpotential of 158 mV for the HER and 50 mA cm^{-2} with an overpotential of 335 mV for the OER in 1.0 M KOH, exhibiting superior electrocatalytic activity and durability. The *in situ* strongly coupled interactions between NiS and Ni contributed to the enhanced electron interactions and lowered charge transfer barrier at the interface, which were beneficial for efficient electrocatalytic reactions. In fact, the interface between active Ni-based sulfides and Ni foam played a key role in generating their excellent electrocatalytic performance because a rational interface could create new electrochemically active sites *via* inducing unsaturated bonding, lattice distortion, and interfacial polarization.

Co-based sulfides also exhibit different phases and compositions, which can greatly affect the electrocatalytic activity and stability toward overall water splitting, with examples such as Co_9S_8 , Co_3S_4 , and CoS_2 . Ma *et al.* synthesized Co_9S_8 , Co_3S_4 , and CoS_2 hollow nanospheres *via* adjusting the molar ratios of S and Co precursors using a facile solution-based strategy.¹¹⁵ The as-prepared CoS_2 with a Co-rich component was superior to Co_9S_8 and Co_3S_4 in 1.0 M KOH, exhibiting overpotentials of 193 mV for the HER and 290 mV for the OER at 10 mA cm^{-2} , and respective Tafel slopes of 57 and 100 mV dec^{-1} . The electrocatalytic activities of these binary Co_x species were determined based on the coordination environments of Co^{n+} active sites, the surface morphology, and the crystallographic structure. As is

commonly known, metal atoms of transitional metal sulfides play important roles in determining electrocatalytic performance. Simultaneously, sulfur vacancies are also crucial in electrocatalytic processes. Water adsorption and dissociation on the low-index (100) facets of Co_9S_8 and Co_3S_4 have been investigated *via* a DFT approach.¹¹⁶ The sulfur vacancies on the (100) facet of Co_9S_8 enhanced the electrocatalytic activity toward water dissociation *via* elevating the energy level of unhybridized Co 3d states to the Fermi level, while having no significant impact on the energetics of the Co_3S_4 (100) facet.

Fe-based sulfides are emerging as a new generation of earth-abundant, low-cost, and highly active alternatives for water electrolysis. However, research into the OER performance of FeS_2 is undertaken much less frequently than research into the HER electrocatalytic activity.^{38,81} Li *et al.* prepared FeS_2/C nanoparticles on Ni foam *via* a facile hydrothermal strategy.¹¹⁷ These FeS_2/C nanoparticles had excellent electrocatalytic activity, long-term durability, and fast charge-transfer kinetics, and they could deliver 10 mA cm^{-2} at overpotentials of 240 and 202 mV for the OER and HER, respectively.

4.2.2 Strategies for enhancing the electrocatalytic performance of non-layered TMS. Various inventive strategies have been developed for optimizing and improving the electrocatalytic performances of non-layered TMS *via* increasing the number of active sites and enhancing the intrinsic activity and electrical conductivity. Typically, synergetic composite designing, strain engineering, and heteroatom doping can significantly enhance the charge transfer kinetics during water electrolysis. Facet engineering, edge engineering, and defect engineering can improve the intrinsic electrocatalytic activities of non-layered TMS *via* exposing more active facets or creating more accessible active sites. In practice, these strategies are not mutually independent and sometimes can be applied simultaneously.

Designing composites is a conventional method for enhancing the charge transfer kinetics and creating new active sites; this can lead to a synergetic effect from each component and trigger the creation of additional interfaces and defects. Active species, like metal-based sulfides, oxides, and carbides, and carbon-based materials, all can be coupled with non-layered TMS to enhance bifunctional electrocatalytic activities. Li *et al.* reported complex polymetallic sulfide systems based on $1\text{T}'\text{-MoS}_2$ and $(\text{Co}, \text{Fe}, \text{Ni})_9\text{S}_8$ as highly active bifunctional nanotube-array electrodes for water splitting, which were denoted as FeCoNi-HNTAs .¹¹⁸ The preparation process and microstructures of FeCoNi-HNTAs are shown in Fig. 14a–c. The electrode only required low overpotentials of 58 and 184 mV at 10 mA cm^{-2} for the HER and OER respectively, while maintaining good long-term stability. Further, when the electrodes were assembled into an electrolyzer, 10 mA cm^{-2} could be delivered at a low cell voltage of 1.429 V and excellent durability was exhibited (Fig. 14d and e). The outstanding electrocatalytic performance was principally thanks to the systematic optimization of the chemical composition and geometric structure. Abundant active sites, the excellent conductivity of metallic $1\text{T}'\text{-MoS}_2$, the synergistic effect of Fe, Co, and Ni ions, and the superaerophobicity of the electrode surface all contributed to the prominent electrocatalytic performance.

The interfaces between non-layered TMS and other materials in composites can bring about the proliferation of active sites. Li *et al.* constructed $\text{FeS}_2/\text{CoS}_2$ nanosheets with abundant defects at their interfaces as efficient bifunctional electrocatalysts.¹¹⁹ Benefiting from the disordered nanosheet interface structure with rich defects, $\text{FeS}_2/\text{CoS}_2$ nanosheets showed excellent electrocatalytic performance, with low overpotentials of 78.2 mV at 10 mA cm^{-2} for the HER and 302 mV at 100 mA cm^{-2} for the OER in 1.0 M KOH. More importantly, the $\text{FeS}_2/\text{CoS}_2$ nanosheets displayed remarkable performance for overall

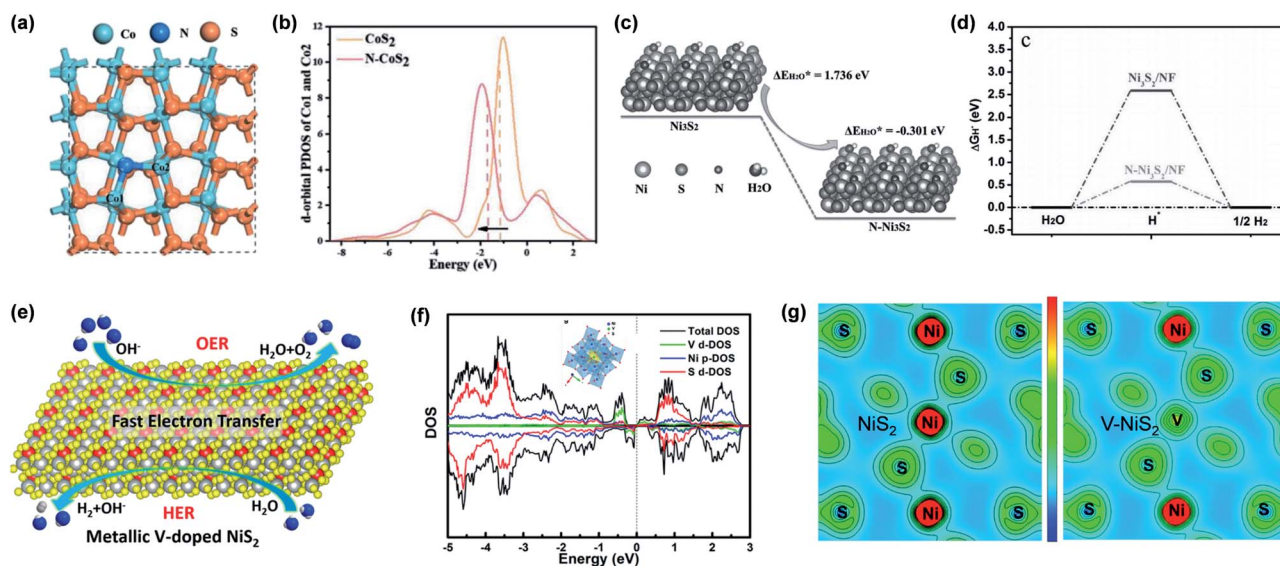


Fig. 15 (a) A top view of the optimized N- CoS_2 (001) structure and (b) the d-orbital partial density of states of Co1 and Co2 from the areas labeled in (a).¹³³ (c) The calculated water adsorption energy and (d) the HER free-energy change for N- Ni_3S_2 and pristine Ni_3S_2 .¹³⁵ (e) A schematic diagram of overall water splitting with V- NiS_2 (with 10% V molar doping). (f) The calculated DOS of V- NiS_2 . (g) Charge-density contour maps of NiS_2 and V- NiS_2 .¹³⁴

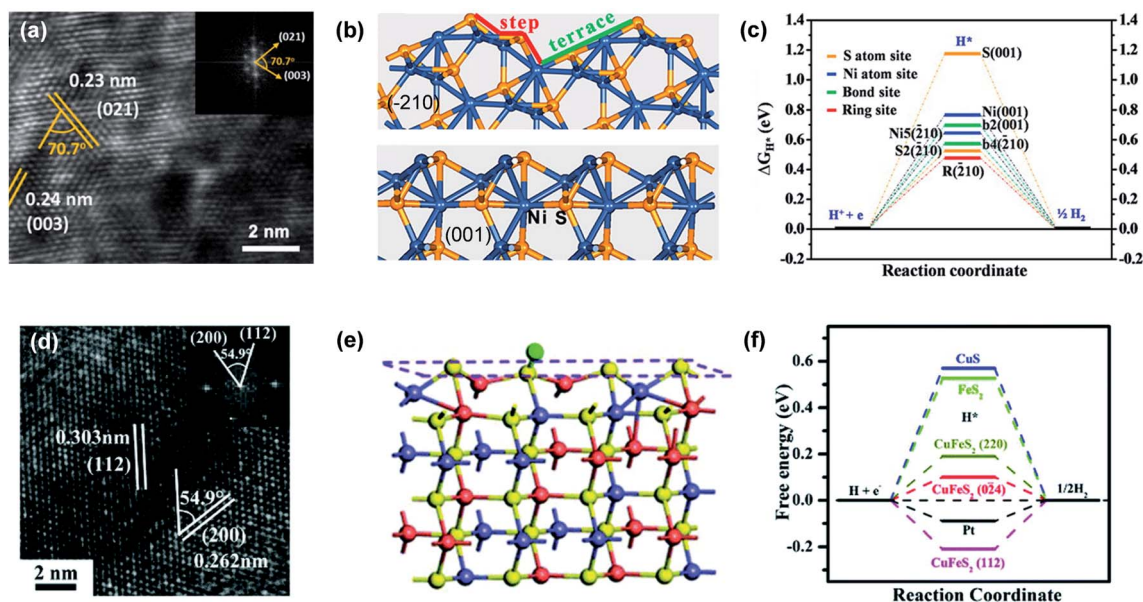


Fig. 16 (a) HRTEM and fast Fourier transform images of Ni_3S_2 . (b) The most stable terminations of the (-210) and (001) surfaces of Ni_3S_2 ; the yellow and blue spheres represent S and Ni atoms, respectively. (c) A calculated free-energy diagram for the HER on the (-210) and (001) surfaces of Ni_3S_2 at equilibrium potential.¹⁴² (d) A HRTEM image of CuFeS_2 nanosheets. (e) A chemisorption model of H on the $(0-24)$ facet of CuFeS_2 nanosheets; the brown, yellow, blue, and green spheres represent Cu, S, Fe, and H atoms, respectively. (f) A calculated free energy diagram for the HER using different materials.¹⁴⁵

water splitting with a cell voltage of 1.47 V required to achieve 10 mA cm^{-2} , and this activity could be maintained for more than 21 h. Zhu *et al.* designed and synthesized a hybrid system involving a cubic Co_9S_8 @ MoS_2 core-shell structure on carbon

nanofibers for overall water splitting, as shown in Fig. 14f-i.¹²⁰ The Co_9S_8 nanoparticle core was surrounded by a shell of fullerene-like MoS_2 layers to form a unique Co_9S_8 @ MoS_2 core-shell structures on carbon nanofibers (CNFs) hybrid system.

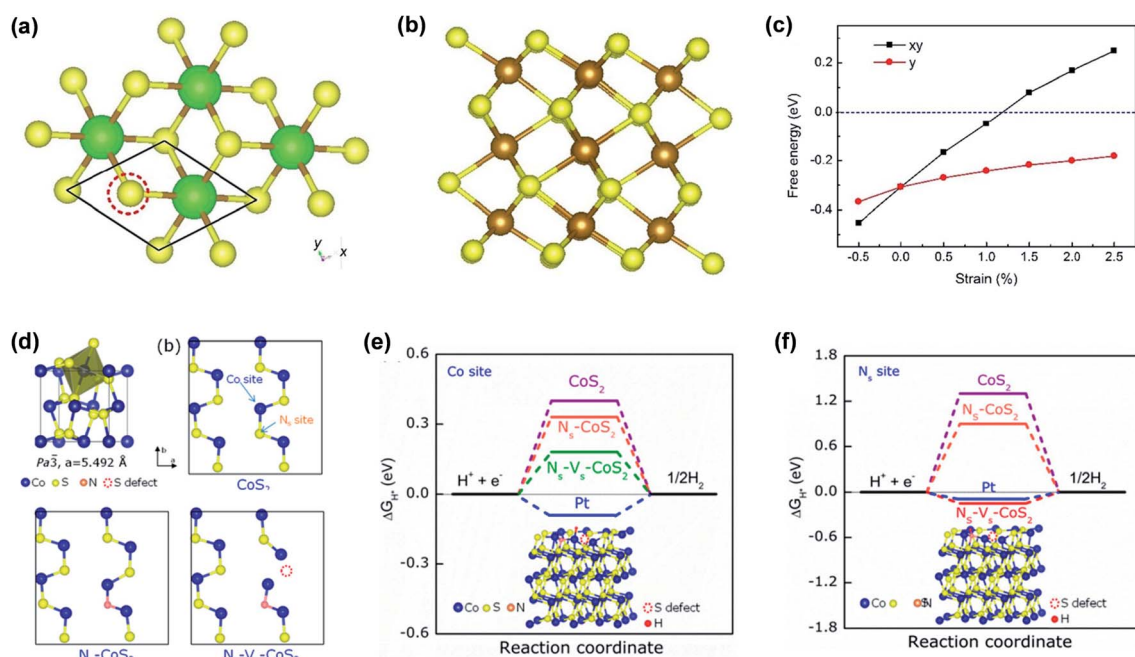


Fig. 17 The charge densities of an optimized (1×1) Fe_3S_4 monolayer (a) without external strain and (b) with 1.0% compressed strain. (c) The Gibbs free energy of hydrogen adsorbed on the surface of a Fe_3S_4 monolayer as a function of the external strain.¹⁴⁹ (d) The crystal structure of CoS_2 and models of surface single-layer atoms in CoS_2 , N-doped CoS_2 ($\text{N}_s\text{-CoS}_2$), and N-doped CoS_2 with a S vacancy ($\text{N}_s\text{-V}_s\text{-CoS}_2$). (e) The HER free energy diagram for Co sites. (f) The HER free energy diagram for N_s sites.¹⁵⁰

Table 2 Summary of non-layered TMS-based bifunctional electrocatalysts for overall water splitting

| TMS-based materials | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential η (mV) | Tafel slope (mV dec ⁻¹) | Cell voltage (V @ mA cm ⁻²) | Stability |
|---|--|---|--------------------------------------|------------------------------|--|---|-----------|
| FeS ₂ /C nanoparticles ¹¹⁷ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 202, 240 | 98, 92 | 1.72 | 5 h |
| FeS nanosheets ¹⁸⁵ | Solvothermal | 1 M KOH | HER/100, OER/10 | 243, 238 | 77, 82.7 | — | 50 h |
| Ni-FeS ₂ (ref. 186) | Electrodeposition | 1 M KOH | HER/10, OER/100 | 157, 255 | 112.5, 58.3 | 1.588 | 11 h |
| Co-FeS ₂ nanospheres ¹⁸⁷ | Solvothermal | 1 M KOH | HER/10, OER/10 | 267, 324 | 58, 50 | 1.60 | 10 h |
| NiS ₂ /FeS ₂ /N-doped carbon nanorods ¹⁸⁸ | Carbonization + solid-phase synthesis | 1 M KOH | HER/10, OER/20 | 172, 231 | 115.64, 44.29 | 1.58 | 80 h |
| FeS/NiS/Ni foam ¹⁸⁹ | Deposition + solvothermal | 1 M KOH | HER/10, OER/10 | 144, 203 | 120, 39 | 1.618 | 10 h |
| CoS ₂ hollow nanospheres ¹¹⁵ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 193, 290 | 100, 57 | 1.54 | 10 h |
| CoS ₂ nanosheets ¹⁹⁰ | CVD | 1 M KOH | HER/10, OER/10 | 90, 220 | 48, 92 | 1.58 | 22 h |
| CoS ₂ nanotube arrays ¹⁴⁵ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 193, 276 | 88, 81 | 1.67 | 20 h |
| CoS _x freestanding sheets ¹⁹¹ | Liquid-phase growth | 1 M KOH | HER/10, OER/10 | 127, 288 | 117, 91 | 1.55 @ 20 | 48 h |
| CoS nanoflake array ¹⁹² | Hydrothermal | 1 M KOH | HER/50, OER/10 | 247, 310 | 114.8, 73.4 | 1.72 | 15 h |
| Co ₉ S ₈ nanosheets ¹⁹³ | Hydrothermal | 0.5 M H ₂ SO ₄ , 1 M KOH | HER/10, OER/10 | 178, 206 | 82, 87 | — | 8 h |
| Co ₃ S ₄ nanosheets ¹⁹⁴ | Solvothermal | 1 M KOH | HER/10, OER/10 | 270, 360 | 124.5, 84.7 | 1.63 | 5 h |
| Co-S films ¹⁵⁶ | Electrodeposition | 1 M KOH | HER/100, OER/100 | 124, 322 | 65.5, 66.4 | 1.79 | 150 h |
| CoS ₂ /graphite foam ¹³¹ | Hydrothermal | 1 M KOH | HER/20, OER/20 | 224, 298 | 148, 82.6 | 1.74 @ 20 | 18 h |
| Co ₃ S ₄ @carbon microflowers ¹⁵² | Hydrothermal | 1 M KOH | HER/10, OER/10 | 140, 250 | 103, 78 | 1.58 | 8 h |
| Mo-Co ₉ S ₈ nanorod array ¹⁹⁵ | Hydrothermal | 1 M KOH | HER/10, OER/50 | 139, 210 | 57, 123 | 1.50 | 20 h |
| Cu-Co ₉ S ₈ nanorod array ¹⁹⁶ | Solid-phase synthesis | 1 M KOH | HER/10, OER/50 | 62, 260 | 76.7, 132.4 | 1.49 | 25 h |
| Cr-Co ₉ S ₈ nanoarrays ¹⁹⁷ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/100 | 120, 313 | 138.8, 10.1 | 1.45 | 12 h |
| Ni-Co ₉ S ₈ nanoarrays ¹⁹⁸ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/100 | 90, 300 | 62.8, 105.1 | 1.45 | 15 h |
| Ni-Co ₃ S ₄ nanowires ¹⁹⁹ | Hydrothermal | 1 M KOH | HER/10, OER/100 | 199, 283 | 91, 65 | 1.70 | 40 h |
| F-CoS ₂ /NF composites ²⁰⁰ | Hydrothermal | 1 M KOH | HER/10, OER/100 | 112, 1596 | 61, 104 | — | 100 h |
| N-CoS ₂ /Ni foam ¹³³ | Hydrothermal + heating | 1 M KOH | HER/10, OER/20 | 28, 200 | 42.6, 55 | 1.50 | 11 h |
| P-CoS ₂ /carbon cloth ²⁰¹ | Solvothermal | 1 M KOH | HER/10, OER/10 | 80, 250 | 98, 90 | 1.56 | 24 h |
| Co _{0.9} S _{0.58} P _{0.42} (ref. 202) | Solid-phase synthesis + phosphorization | 1 M KOH | HER/10, OER/10 | 140, 266 | 78, 48 | 1.59 | 30 h |
| TiO ₂ @Co ₉ S ₈ core-branch arrays ²⁰³ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 139, 240 | 65, 55 | 1.56 | 30 h |
| P-CoMoS/carbon cloth ²⁰⁴ | Hydrothermal + phosphorization | 1 M KOH | HER/10, OER/10 | 66, 260 | 60.1, 72.2 | 1.54 | 24 h |
| CoS@few-layer graphene ²⁰⁵ | High-temperature treatment | 1 M KOH | HER/10, OER/10 | 118, 350 | 63, 70.9 | 1.77 | 10 h |
| CoS ₂ @N-doped graphene ²⁰⁶ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 204, 243 | 108, 51.8 | 1.53 | 12 h |

Table 2 (Contd.)

| TMS-based materials | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential (mV) | η Tafel slope (mV dec ⁻¹) | Cell voltage | | Stability |
|---|--|--------------------------------------|--------------------------------------|-----------------------|--|--------------|------------------|-----------|
| | | | | | | (V @ mA) | cm ⁻² | |
| N-, O-, S-tridoped carbon-encapsulated Co ₉ S ₈ (ref. 207) | Pyrolysis | 1 M KOH | HER/10, OER/10 | 320, 340 | 105, 68 | 1.60 | 10 h | |
| | Polyol refluxing + solid-phase synthesis + calcination | 1 M KOH | HER/10, OER/10 | 332.4, 266 | 131.4, 75.5 | — | 20 h | |
| Co ₉ S ₈ /N ₁ S-rGO ¹³⁰ | Self-catalysis + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 112, 157 | 104.9, 76.1 | 1.633 | 20 h | |
| S ₁ N-CNTs/CoS ₂ @Co nanoparticles ²⁰⁸ | Pyrolysis | 0.5 M H ₂ SO ₄ | HER/10, OER/10 | 74, — | 69.3, — | — | — | |
| | | 1.0 M phosphate solution | HER/10, OER/10 | 121, — | 106.4, — | — | — | |
| | | 1 M KOH | HER/10, OER/10 | 89, 293 | 86.7, 59.7 | 1.61 | 20 h | |
| Co ₉ S ₈ /MnS/Ni foam ²⁰⁹ | Solvothermal | 1 M KOH | HER/100, OER/100 | 217, 298 | 48.2, 43.9 | 1.47 | 20 h | |
| CoNi ₂ S ₄ @CoS ₂ /Ni foam ²¹⁰ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 173, 259 | 51, 45 | — | 16 h | |
| Co ₉ S ₄ P ₄ pentlandite ¹⁵⁹ | Etching + CVD | 1.0 M PBS | HER/10, OER/25.9 | 87, 570 | 51, 106 | 1.67 | 24 h | |
| Zn _{0.3} Co _{2.7} S ₃ P nanoparticles ²¹¹ | Solvothermal + phosphorization | 1 M KOH | HER/10, OER/10 | 188, 261 | 76.1, 56.2 | 1.70 | 10 h | |
| CuCo ₂ S ₄ nanowire arrays ²¹² | Hydrothermal | 1 M KOH | HER/10, OER/100 | 65, 310 | 42, 49 | 1.65 @ 100 | 50 h | |
| CuCo ₂ S ₄ (ref. 213) | Hydrothermal | 1 M KOH | HER/10, OER/20 | 158, 290 | 113, — | 1.66 | 24 h | |
| CuCo ₂ S ₄ @Ni(OH) ₂ nanorods ²¹⁴ | Hydrothermal + electrodeposition | 1 M KOH | HER/10, OER/100 | 117, 250 | 170, 103.4 | 1.47 | 20 h | |
| FeCo ₂ S ₄ nanosheet arrays ²¹⁵ | Hydrothermal | 1 M KOH | HER/10, OER/50 | 132, 270 | 164, 59 | 1.56 | 20 h | |
| FeCo ₂ S ₄ nanosheet arrays ¹³⁸ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/20 | 131, 259 | 52, 173 | 1.541 | 12 h | |
| Fe-Co-S nanosheets ²¹⁶ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 68, 208 | 114, 37 | 1.44 | 80 h | |
| NiCo ₂ S ₄ nanowire array ²¹⁷ | Wet-chemistry solid-phase synthesis | 1 M KOH | HER/50, OER/50 | 266, 300 | 130.5, 90.9 | 1.66 | 20 h | |
| NiCo ₂ S ₄ nanowire arrays ¹³⁷ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 210, 260 | 58.9, 40.1 | 1.63 | 50 h | |
| NiCo ₂ S ₄ nanoflakes ²¹⁸ | Reverse microemulsion + solid-phase synthesis | 1 M KOH | HER/10, OER/100 | 169, 319 | 97.1, 53.3 | 1.61 | 70 h | |
| NiCo ₂ S ₄ nanowires ²¹⁹ | Hydrothermal | 1 M KOH | HER/10, OER/40 | 191, 256 | 116.9, 67.7 | 1.59 | 12 h | |
| Ni _{4.3} Co _{4.7} S ₈ (ref. 220) | Hydrothermal | 1 M KOH | HER/10, OER/20 | 148, 133.8 | 90, 194.2 | 1.67 @ 50 | — | |
| (Ni _{0.33} Co _{0.67})S ₂ nanowires/carbon cloth ²²¹ | Hydrothermal | 1 M KOH | HER/100, OER/100 | 334, 216 | 127, 78 | 1.57 | 30 h | |
| Ni-Mo-S nanowires ²²² | Electroplating + solid-phase synthesis | 1 M KOH | HER/100, OER/100 | 290, 390 | 103, 75 | 2 @ 100 | 12 h | |
| NiFeMoS nanorods ²²³ | Electrodeposition + hydrothermal | 1 M KOH | HER/10, OER/150 | 100, 280 | 121, 69 | 1.52 @ 100 | 10 h | |
| Ni ₈ V ₂ (Mo ₃ S ₄) ₁₁ /Ni-Co foam ²²⁴ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 129, 330 | 136, 183 | — | 25 h | |
| NiCo/NiCo ₂ S ₄ @NiCo arrays ²²⁵ | Hydrothermal + electrodeposition | 1 M KOH | HER/10, OER/100 | 132, 294 | 58.2, 59.6 | 1.55 | 27 h | |
| FeCo ₂ S ₄ -NiCo ₂ S ₄ nanosheet arrays ¹⁴⁰ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 150, 230 | 38, 65 | 1.51 | 10 h | |
| Cu@CoS _x /copper foam ²²⁶ | Chemical deposition | 1 M KOH | HER/10, OER/10 | 134, 160 | 61, — | 1.50 | 200 h | |
| Ni-Fe-Co-S nanosheets ²²⁷ | Electrodeposition | 1 M KOH | HER/10, OER/10 | 106, 207 | 95, 63 | 1.54 | 10 h | |
| Ni-Co-S nanosheet arrays ²²⁸ | Annealing + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 110, 270 | 56, 80 | 1.62 | 12 h | |

Table 2 (Contd.)

| TMS-based materials | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential η (mV) | Cell voltage | | Stability |
|--|---|--|--------------------------------------|------------------------------|--|-------------------------------|-----------|
| | | | | | Tafel slope (mV dec ⁻¹) | (V @ mA cm ⁻²) | |
| P-(Ni, Fe) ₃ S ₂ nanosheet arrays ²²⁹ | Phosphorization + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 98, 196 | 88, 30 | 1.54 | 15 h |
| (Ni-Fe) _x /NiFe(OH) _y hollow microtubes/spheres ²³⁰ | Electrodeposition | 1 M KOH | HER/100, OER/100 | 124, 290 | 68, 58 | 1.46 | 50 h |
| Co(OH) ₂ /Ni-Co-S nanotube arrays ²³¹ | Hydrothermal | 1 M KOH | HER/100, OER/100 | 254, 340 | 88, 64 | 1.62 | 20 h |
| Co ₉ S ₈ /Ni ₃ Se ₂ /graphene ²³² | Hydrothermal + solvothermal | 1 M KOH | HER/20, OER/10 | 170, — | 83, — | 1.62 | 10 h |
| Ni-Mo-S@Co ₃ O ₄ /carbon fibers ²³³ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 85, 275 | 50, 109 | 1.57 | 50 h |
| N-carbon coated NiCo ₂ S ₄ hollow nanotubes ²³⁴ | Solvothermal | 1 M KOH | HER/100, OER/100 | 295, 330 | 89.8, 86.8 | 1.60 | 15 h |
| rGO/(Ni _x Mn _y Co _z) ₃ S ₄ nanocomposites ²³⁵ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 151, 320 | 52, 58 | 1.56 @ 20 | 24 h |
| Nanoporous (003)-Ni ₃ S ₂ (ref. 50) | Anodization + solid-phase synthesis | 1 M NaOH | HER/10, OER/10 | 135, 175 | 75.7, 101.2 | 1.611 | 30 h |
| (-210)-Ni ₃ S ₂ nanosheet arrays ¹⁴² | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 223, 260 | —, — | — | 200 h |
| NiS microsphere film ¹¹⁴ | Solid-phase synthesis | 1 M KOH | HER/20, OER/50 | 158, 335 | 83, 89 | 1.64 | 35 h |
| Ni ₃ S ₂ superstructures ¹¹² | Chemical etching | 1 M KOH | HER/10, OER/100 | 135, 320 | 75, 59 | 1.59 | 50 h |
| Ni ₃ S ₂ nanocrystals ¹¹¹ | Ambient-pressure polyol solution | 1 M KOH | HER/10, OER/10 | 112, 295 | 67, 52 | 1.63 | 108 h |
| Ni ₃ S ₂ nanorods ²³⁶ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 82, 310 | 73.8, 80.1 | 1.61 | 30 h |
| Ni ₃ S ₂ superstructures/Ni foam ²³⁷ | Chemical etching | 1 M KOH | HER/10, OER/10 | 182, 340 | 89, 150 | — | 12 h |
| Ni ₃ S ₂ /Ni foam ²³⁸ | Annealing | 1 M KOH | HER/10, OER/10 | 131, 312 | 96, 111 | 1.68 | 14 h |
| Coral-like Ni ₃ S ₂ /Ni foam ¹¹³ | Electrodeposition | 1 M KOH | HER/10, OER/10 | 89, 242 | 85, 74 | 1.577 | 40 h |
| Ni ₃ S ₂ /Ni foam ²³⁹ | Solvothermal | 1 M KOH | HER/10, OER/10 | 123, 222 | 91, 60 | 1.59 | 20 h |
| NiS ₂ nanospheres ¹¹⁰ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 147, 241 | 105, 65 | 1.66 | 40 h |
| NiS ₂ nanowires/carbon fiber paper ²⁴⁰ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 165, 246 | 134, 94.5 | 1.59 | 20 h |
| NiS ₂ microspheres and NiS microspheres ⁶⁷ | Hydrothermal + annealing | 1 M KOH | HER/10, OER/10 | 148, 320 | 79, 59 | 1.58 | 12 h |
| Ni ₇ S ₆ foam ²⁴¹ | Annealing | 0.5 M H ₂ SO ₄ , 0.1 M KOH | HER/10, OER/10 | 70, 140 | 63, 119 | —, 1.51 | 24 h |
| NiS/Ni coupled interface ¹⁴⁶ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 161, 301 | 74, 46 | — | 18.6 h |
| Ni/NiS/N-doped mesoporous carbon ²⁴² | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 70, 337 | 45, 52 | 1.61 | 25 h |
| Ni(S _{0.5} Se _{0.5}) ₂ hollow/porous spheres ²⁴³ | Hydrothermal + selenization | 1 M PBS | HER/10, OER/100 | 124, 501 | 81, 94 | 1.87 | 12 h |
| N-Ni ₃ S ₂ (ref. 135) | Annealing | 1 M KOH | HER/10, OER/100 | 110, 330 | —, 70 | 1.48 | 8 h |
| Sn-Ni ₃ S ₂ nanosheets ²⁴⁴ | Hydrothermal | 1 M KOH | HER/100, OER/100 | 170, 270 | 55.6, 52.7 | 1.46 | 60 h |
| Mo-Ni ₃ S ₂ nanorods ¹⁵³ | Hydrothermal | 1 M KOH | HER/100, OER/100 | 278, 180 | 72.9, 45.5 | 1.53 | 15 h |
| Mo-Ni ₃ S ₂ nanosheets ²⁴⁵ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 212, 260 | 98, 85 | 1.67 | 13 h |
| Mo-NiS heterostructures ²⁴⁶ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/50 | 147.6, 390 | 88.1, 185 | 1.92 | 18 h |
| Mo/Mn-Ni _x S _y /Ni foam ²⁴⁷ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/50 | 144, 162 | 91, 110 | 1.49 | 24 h |

Table 2 (Contd.)

| TMS-based materials | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential (mV) | η Tafel slope (mV dec ⁻¹) | Cell voltage | | Stability |
|--|---|-------------|--------------------------------------|-----------------------|---|-------------------------------|-------|-----------|
| | | | | | | (V @ mA cm ⁻²) | | |
| Ni _{0.7} Fe _{0.3} S ₂ microflowers ¹³⁹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 155, 198 | 109, 56 | 1.625 | 14 h | |
| Fe-Ni ₃ S ₂ nanowires ²⁴⁸ | Solvothermal | 1 M KOH | HER/10, OER/200 | 109, 223 | 49.5, 55.7 | 1.53 | 14 h | |
| Fe-Ni ₃ S ₂ nanosheet arrays ⁸ | Hydrothermal + ion exchange | 1 M KOH | HER/10, OER/10 | 47, 214 | 95, 42 | 1.54 | 10 h | |
| Fe _{11.1%} -Ni ₃ S ₂ /Ni foam ²⁴⁹ | Hydrothermal + wet-chemistry conversion | 1 M KOH | HER/50, OER/100 | 203, 252 | 89, 61.7 | 1.60 | 42 h | |
| Co-Ni ₃ S ₂ nanoarrays ²⁵⁰ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 240, 120 | 145, 38.4 | 1.69 @ 50 | 10 h | |
| V-NiS ₂ nanosheets ¹³⁴ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 110, 290 | 90, 45 | 1.56 | 20 h | |
| V-Ni ₃ S ₂ nanorod array ²⁵¹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 133, 148 | 70, 42 | 1.421 | 60 h | |
| V-Ni ₃ S ₂ @NiO/Ni foam ²⁵² | Hydrothermal | 1 M KOH | HER/10, OER/10 | 112, 170 | 69, 98.8 | 1.52 | 55 h | |
| (-210)-Ni ₃ S ₂ branch arrays ²⁵³ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 112, 220 | 69, — | 1.58 | 10 h | |
| MoO _x /Ni ₃ S ₂ microspheres ¹²⁵ | Solid-phase synthesis + hydrothermal | 1 M KOH | HER/10, OER/10 | 106, 136 | 90, 50 | 1.45 | 200 h | |
| Ni ₃ S ₂ /VO ₂ core/shell nanoarray ²⁵⁴ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 100, 150 | 114, 47 | 1.42 | 15 h | |
| Ni ₃ N/Ni ₃ S ₂ (ref. 255) | Calcination | 1 M KOH | HER/100, OER/100 | 197, 404 | 58.8, 112 | — | 12 h | |
| Ni ₃ S ₂ /MnS nanosheets ²⁵⁶ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 116, 228 | 41, 46 | 1.54 | 50 h | |
| Se-MnS/NiS heterojunctions ²⁵⁷ | Hydrothermal chemical- deposition | 1 M KOH | HER/10, OER/10 | 56, 211 | 55, 50 | 1.47 | 48 h | |
| CdS/Ni ₃ S ₂ nanosheet flowers ²⁵⁸ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 121, 151 | 110, 174 | — | 12 h | |
| Ni _x Co _{3-x} S ₄ /Ni ₃ S ₂ /Ni foam ²⁵⁹ | Cation exchange | 1 M KOH | HER/10, OER/10 | 136, 160 | 107, 95 | 1.53 | 200 h | |
| CoMoS ₄ /Ni ₃ S ₂ nanostructures ¹⁵⁸ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 158, 200 | 169, 63 | 1.568 | 10 h | |
| N-NiMoO ₄ /NiS ₂ nanowires/ nanosheets ²⁶⁰ | Solid-phase synthesis | 1 M KOH | HER/10, OER/10 | 99, 283 | 74.2, 44.3 | 1.60 | 20 h | |
| Carbon dots/NiCo ₂ S ₄ /Ni ₃ S ₂ (ref. 157) | Hydrothermal + cation exchange | 1 M KOH | HER/10, OER/10 | 127, 116 | 148, 99 | 1.50 | 12 h | |
| Mo _(1-x) W _x S ₂ /Ni ₃ S ₂ heterostructures ²⁶¹ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 98, 285 | 92, 98 | 1.62 | 50 h | |
| NiS-CoS nanorod arrays ²⁶² | Electrodeposition + hydrothermal | 1 M KOH | HER/10, OER/10 | 102, 170 | 92, 71 | 1.47 | 12 h | |
| NiS ₂ /CoS ₂ microstructures ¹²² | Solid-phase synthesis | 1 M KOH | HER/10, OER/20 | 165, 310 | 72, 78 | 1.61 | 24 h | |
| Co ₉ S ₈ /Ni ₃ S ₂ nanowire arrays ²⁶³ | Hydrothermal + vulcanization | 1 M KOH | HER/10, OER/10 | 128, 227 | 97.6, 46.5 | 1.64 | 9 h | |
| Co ₉ S ₈ @Ni(OH) ₂ nanorods ²⁶⁴ | Hydrothermal + electrodeposition | 1 M KOH | HER/10, OER/60 | 100, 250 | 143.5, 273.4 | 1.55 | 10 h | |
| CoS _x /Ni ₃ S ₂ heterostructures ²⁶⁵ | Hydrothermal | 1 M KOH | HER/10, OER/20 | 204, 280 | 113.1, 92.7 | 1.572 | 30 h | |
| Ni ₃ S ₂ /Co ₉ S ₈ arrays ¹²¹ | Solid-phase synthesis | 1 M KOH | HER/100, OER/100 | 269, 340 | 98, 66 | 1.55 | 12 h | |
| CoS/Ni ₃ S ₂ -FeS nanopetals ²⁶⁶ | Electrodeposition + solid- phase synthesis | 1 M KOH | HER/10, OER/10 | 75, 136 | 57, 51 | — | 24 h | |
| FeS ₂ /CoS ₂ interface nanosheets ¹¹⁹ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/100 | 78.2, 302 | 44, 42 | 1.47 | 21 h | |
| Cu ₂ S-Ni ₃ S ₂ array ²⁶⁷ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 149, 329 | 75.89, 44.11 | 1.77 | 100 h | |

Table 2 (Contd.)

| TMS-based materials | Preparation method | Electrolyte | Reaction/J (mA cm ⁻²) | Overpotential η (mV) | Tafel slope (mV dec ⁻¹) | Cell voltage | | Stability |
|---|---|-------------|--------------------------------------|------------------------------|--|----------------|---------------------|-----------|
| | | | | | | (V @ mA) | (cm ⁻²) | |
| Ni ₃ S ₂ /VS ₄ nanohorn array ²⁶⁸ | Solvothermal | 1 M KOH | HER/10, OER/50 | 177, 317 | 139, 43 | 1.57 | | 70 h |
| NiS/NiS ₂ (ref. 123) | Thermal decomposition | 1 M KOH | HER/100, OER/100 | 248, 416 | 95.1, 156.5 | 1.62 | | 36 h |
| NiS–NiS ₂ –Ni ₃ S ₂ /Ni foam ¹²⁴ | Solid-phase synthesis | 1 M KOH | HER/20, OER/20 | 137, 143 | 87, 80 | 1.46 | | 14 h |
| Mn-doped Cu _{7.2} S ₄ @NiS ₂ @NiS ²⁶⁹ | Hydrothermal + solid-phase synthesis | 1 M KOH | HER/10, OER/50 | 87, 252 | 43.7, 130.1 | 1.514 | | 46 h |
| Mo–Ni ₃ S ₂ /Ni _x P _y hollow nanorods ²⁷⁰ | Solvothermal | 1 M KOH | HER/10, OER/50 | 109, 238 | 68.4, 60.6 | 1.46 | | 72 h |
| Ni ₂ P/Ni ₃ S ₂ heteronanoflake arrays ²⁷¹ | Hydrothermal + phosphorization | 1 M KOH | HER/10, OER/10 | 80, 210 | 65, 62 | 1.50 | | 36 h |
| NiS/Ni ₂ P heterostructures ²⁷² | Hydrothermal + solid-phase synthesis + phosphorization | 1 M KOH | HER/20, OER/20 | 111, 265 | 78.1, 41.3 | 1.67 | | 10 h |
| N–Ni ₃ S ₂ /VS ₂ nanosheets ²⁷³ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 151, 227 | 107.5, 70.5 | 1.648 | | 20 h |
| Ni(OH) ₂ /Ni ₃ S ₂ nanosheet arrays ²⁷⁴ | Co-precipitation + solid-phase synthesis | 1 M KOH | HER/20, OER/20 | 270, 211 | 129, 152.7 | 1.57 | | 10 h |
| NiS/Ni ₂ P@Ni(OH) ₂ (ref. 275) | Electrodeposition | 1 M KOH | HER/10, OER/10 | 120, 219 | 71, 82 | 1.58 | | 160 h |
| NiFe-LDH@Ni ₃ S ₂ nanosheet arrays ²⁷⁶ | Hydrothermal | 1 M KOH | HER/20, OER/50 | 197, 230 | 99, 29 | 1.65 | | 12 h |
| Se–(NiCo)S _x /(OH) _x nanosheets ²⁷⁷ | Solvothermal | 1 M KOH | HER/10, OER/10 | 103, 155 | 87.3, 33.9 | 1.60 | | 66 h |
| N-CNTs/NiS ₂ @Mo ₂ C ¹²⁷ | Calcination + CVD | 1 M KOH | HER/10, OER/10 | 227, 320 | 114.6, 77.5 | 1.52 | | 24 h |
| Ni ₃ S ₂ /N-CNTs ¹²⁹ | Solid-phase synthesis | 1 M KOH | HER/50, OER/50 | 135, 380 | 58, 77 | 1.721 @ 100 | | 50 h |
| Ni ₃ S ₂ @graphite foam@Ni foam ²⁷⁸ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 99, 240 | 98.2, 62.4 | 1.63 | | 30 h |
| N-doped graphene QDs/Ni ₃ S ₂ nanosheets ¹²⁸ | Hydrothermal | 1 M KOH | HER/10, OER/10 | 218, 216 | 89, 95.5 | 1.58 | | 17 h |
| 3D graphene–Au–Ni ₃ S ₂ (ref. 279) | Electrodeposition | 1 M KOH | HER/10, OER/91.2 | 140, 370 | 93, 148 | 1.63 | | 19 h |

When compared with pure MoS₂/CNFs and Co₉S₈/CNFs nanostructures, the hybrid system exhibited impressive enhanced HER and OER activities, which were attributed to a synergetic effect between Co and Mo in the proposed localized nano-interface region between Co₉S₈ and MoS₂. The nano-interfaces could generate strong electron transfer between Mo and Co through intermediate S atoms bonded to both, which led to the promising improved electrocatalytic activity. Various composites based on non-layered TMS for overall water splitting have been exploited due to the diverse types of these materials, such as Ni₃S₂/Co₉S₈ arrays,¹²¹ NiS₂/CoS₂ nanostructures,¹²² interwoven NiS/NiS₂ structures,¹²³ and multiphase NiS–NiS₂–Ni₃S₂.¹²⁴

Metal-based oxides could also be applied as effective enhancers. Wu *et al.* reported ultrathin nanosheet-built, hollow MoO_x/Ni₃S₂ microspheres on Ni foam that possess prominent electrocatalytic activity toward both the OER and HER.¹²⁵ The unique structural features of the composite promoted the exposure of high-density electrocatalytically active sites, requiring an impressive low cell voltage of 1.45 V to reach 10 mA

cm⁻² with remarkable durability for more than 100 h in an alkaline electrolyzer. Furthermore, metal-based carbides, such as Mo₂C with good electrical conductivity and a Pt-like d-orbital electronic structure, have been extensively studied for constructing composites with non-layered TMS, for example, porous Co₉S₈/N, S-doped carbon@Mo₂C¹²⁶ and a NiS₂/carbon nanotubes@Mo₂C bifunctional electrode.¹²⁷

Carbon materials with different dimensions are another good choice for building composites based on non-layered TMS, like 0D graphene quantum dots, 1D carbon nanotubes, 2D graphene, and 3D graphite foam. Lv *et al.* fabricated nitrogen-doped graphene quantum dots (NGQDs) and Ni₃S₂ nanosheets on Ni foam (Ni₃S₂–NGQDs/NF).¹²⁸ Self-supported Ni₃S₂–NGQDs/NF electrodes delivered 10 mA cm⁻² at overpotentials of 216 and 218 mV for the OER and HER, respectively, and required a low voltage of 1.58 V at 10 mA cm⁻² in an alkali electrolyzer. The experimental results and theoretical calculations proved that the excellent performance arose from synergistic effects between the constructed active interfaces of Ni₃S₂ and NGQDs; in other words, NGQDs could effectively regulate

the electrocatalytic activity of Ni_3S_2 . Analogously, Guo *et al.* synthesized metal sulfide (Ni_3S_2 , Co_9S_8 , or FeS) foams *via* the *in situ* conversion of corresponding metal (Ni, Co, or Fe) foams, and these then acted as templates for the surface growth of N-doped carbon nanotube (CNT) arrays for fabricating metal sulfide/CNT foam composites.¹²⁹ Of these, Ni_3S_2 /CNT foam served as the most effective electrode for overall water splitting with voltages of 1.5 and 1.72 V at 10 and 100 mA cm^{-2} , respectively. Ni_3S_2 foam could provide abundant active sites and excellent conductivity; meanwhile, the CNT array coating could supply special superaerophobic geometry and protect the foam from electro-corrosion, indicating that tuning the surface geometry was an efficacious way to promote the electrocatalytic performance. Liu *et al.* grew vertically aligned ultrathin Co_9S_8 nanosheets *in situ* on N and S co-doped reduced graphene oxide ($\text{Co}_9\text{S}_8/\text{N,S-rGO}$) as an efficient electrocatalyst for water electrolysis.¹³⁰ This unique hierarchical architecture was adopted to promote electron transport and fully expose active sites. $\text{Co}_9\text{S}_8/\text{N,S-rGO}$ showed highly efficient and stable electrocatalytic performance for the OER and moderate HER activity in alkaline media, benefitting from a synergistic catalytic effect between Co_9S_8 nanosheets and N,S/rGO. Tong *et al.* prepared self-standing cobalt-disulfide/graphite foam (CoS_2/GF) composite films *via* a hydrothermal approach.¹³¹ This 3D porous CoS_2/GF electrode exhibited ultrahigh electrocatalytic activities for both the HER and OER, delivering a current density of 20 mA cm^{-2} at a cell voltage of 1.74 V in an alkaline electrolyzer. The excellent performances of CoS_2/GF composite films were mainly attributed to the following factors: a strong synergistic effect between CoS_2 and GF; the high electrical conductivity of GF, which could facilitate charge transfer during water electrolysis; the porous inter-layer structure of GF, which could provide effective pathways for releasing H_2 and O_2 bubbles; and the large surface area and numerous active sites of CoS_2/GF composite electrodes.

Heteroatom doping plays a pivotal role in enhancing the electrocatalytic performances of non-layered TMS *via* regulating electronic structures to allow faster electron transfer and to optimize the binding energy of intermediates; strategies include substitutional doping and surface charge transfer doping. The substitutional doping of heteroatoms has attracted widespread attention in recent years, for example, the nitrogen doping of CoS_2 could change the electronic density and minimize the adsorption free energy of materials.¹³² Yao *et al.* synthesized nitrogen-doped 3D dandelion-flower-like CoS_2 structures directly grown on Ni foam (N- CoS_2/NF).¹³³ Benefiting from the unique 3D architecture and optimized hydrogen and water adsorption free energies due to N doping, an N- CoS_2/NF electrode showed remarkable electrocatalytic performance, requiring only a low overpotential of 28 mV to gain 10 mA cm^{-2} for the HER and an overpotential of 200 mV at 20 mA cm^{-2} for the OER in alkaline solution (Fig. 15a and b). In fact, non-noble metal atoms are always applied as dopants due to the merits of high conductivity and low cost. Liu *et al.* reported pyrite NiS_2 nanosheets doped with vanadium heteroatoms, with the conversion of the electronic structure from a typical semi-conducting form to one showing metallic characteristics.¹³⁴ This interesting electronic structure reconfiguration of NiS_2 was

rooted in electron transfer from doped V sites to Ni sites, and it enabled Ni sites to obtain more electrons (Fig. 15e–g). Consequently, the metallic V-doped NiS_2 nanosheets (with 10% V molar doping) displayed outstanding electrocatalytic performance, with overpotentials of 290 and 110 mV for the OER and HER, respectively, at 10 mA cm^{-2} in 1 M KOH.

Ternary and quaternary non-layered TMS with higher conductivity and richer redox reactions always exhibit greatly enhanced electrocatalytic performance, with examples such as Ni–Co–S, Fe–Co–S, and Ni–Fe–S systems.¹³⁶ Sivanantham *et al.* directly grew NiCo_2S_4 nanowire arrays on the surface of 3D Ni foam (NiCo_2S_4 NW/NF) *via* a pressurized hydrothermal method.¹³⁷ Benefiting from a high surface area, a well-distributed nanowire array structure, and enhanced charge transport due to the presence of mixed metals, this self-supported NiCo_2S_4 NW/NF electrode exhibited superior electrocatalytic performance, with low overpotentials of 260 and 210 mV at 10 mA cm^{-2} for the OER and HER, respectively, in 1 M KOH. Gong *et al.* synthesized FeCo_2S_4 nanosheets that were grown *in situ* on Ni foam ($\text{FeCo}_2\text{S}_4/\text{NF}$) using a facile hydrothermal sulfurization method.¹³⁸ The self-supported $\text{FeCo}_2\text{S}_4/\text{NF}$ electrodes delivered 10 mA cm^{-2} , only requiring a small cell voltage of 1.541 V in a two-electrode alkaline electrolyzer. Furthermore, Yu *et al.* prepared hierarchical porous ternary nickel-iron sulfide ($\text{Ni}_{0.7}\text{Fe}_{0.3}\text{S}_2$) microflowers on Ni foam *via* a traditional two-step method.¹³⁹ Owing to the special 3D morphology and strong electron interactions between Ni, Fe, and S atoms, the as-synthesized electrocatalyst exhibited enhanced activity and durability for overall water splitting. The incorporation of Fe not only altered the morphology of sulfides from vertically aligned nanoplates to 3D hierarchical microflowers, but it also tuned the electronic structure of NiS_2 and further optimized the binding energies of reactive intermediates. Furthermore, Li *et al.* reported hierarchical trimetallic sulfide FeCo_2S_4 – NiCo_2S_4 nanosheet arrays supported on Ti mesh as efficient 3D bifunctional electrocatalysts for overall water splitting.¹⁴⁰ This trimetallic sulfide electrode showed superior electrocatalytic performance, with low overpotentials of 230 and 150 mV at 10 mA cm^{-2} for the OER and HER, respectively, which could be attributed to the fully exposed active sites, fast charge transfer, and synergistic effects between multiple components.

Surface charge transfer doping with heteroatoms can be used to modify the electronic structures and morphologies of non-layered TMS simultaneously. Chen *et al.* developed 3D N-anion-decorated Ni_3S_2 on Ni foam (N- $\text{Ni}_3\text{S}_2/\text{NF}$) prepared *via* a one-step calcination route for both the OER and HER.¹³⁵ Remarkably, the as-prepared N- $\text{Ni}_3\text{S}_2/\text{NF}$ electrodes displayed extremely high electrocatalytic activity, long-term stability, and favorable reaction kinetics for both OER and HER processes. Low overpotentials of 330 and 110 mV were needed to reach 100 and 10 mA cm^{-2} for the OER and HER, respectively, in 1.0 M KOH, and a very low cell voltage of 1.48 V was required to deliver a current density of 10 mA cm^{-2} in an overall water-splitting device. The introduction of N anions notably modified the electronic structure and morphology of Ni_3S_2 , leading to fully exposed active sites, enhanced electrical conductivity, and

optimized hydrogen and water adsorption energies (Fig. 15c and d). Wang *et al.* synthesized metallic Co_9S_8 decorated with single-atomic Mo as an extremely efficient bifunctional electrocatalyst under various pH conditions.¹⁴¹ The superior electrocatalytic performance of Mo- Co_9S_8 could be attributed to the following factors: synergistic effects between atomically dispersed Mo and high oxidation state Co led to greatly increased activity and durability; the increased active surface area and superaerophobic interface could facilitate gas release; and the regulation of the electronic structure of Co_9S_8 *via* the immobilization of single-site Mo species on its surface could enhance the electrical conductivity, with a higher density electron cloud and altered binding energies of adsorbed intermediate species.

Facet engineering is a unique method for increasing the intrinsic electrocatalytic activity of non-layered TMS *via* exposing more highly active facets. As is widely known, transition metals in non-layered TMS play a crucial role in determining the electrochemical performance, and they are always responsible for the good electrocatalytic activity. On one hand, transition metal cations on exposed low-index facets usually involve an unsaturated coordination environment, which would possess high electrochemical activity. For example, the low-index (100) facet of FeS_2 is terminated with nonpolar [S-Fe-S] repeated along the regular facet direction, while Fe cations with five-fold coordination are located in a square pyramidal environment, as seen in Fig. 3d.³⁸

On the other hand, the electrocatalytic activities of high-index facets with special structures, such as an asymmetric zigzag structure, will be higher than those of low-index facets in non-layered TMS. Feng *et al.* reported that the exposed (-210) high-index facet with an asymmetric zigzag structure in Ni_3S_2 nanosheet arrays contributed to the remarkable catalytic activity, which was much higher than the (001) low-index facet in Ni_3S_2 .¹⁴² The as-prepared Ni_3S_2 nanosheet arrays with the (-210) high-index facet were synthesized *via* the direct sulfuration of Ni foam in a hydrothermal system and exhibited efficient and stable electrocatalytic activity toward both the HER and OER. DFT computations also elucidated that the HER and OER activities of the (-210) surface were better than those of the low-index (001) surface (Fig. 16a-c). Similarly, Li *et al.* synthesized ultrathin metallic CuFeS_2 nanosheets with abundant exposed high-index (0-24) facets.¹⁴³ Theoretical calculations and experimental results confirmed that the CuFeS_2 nanosheets with exposed (0-24) high-index facets exhibited excellent electrocatalytic HER activity, benefitting from high electrical conductivity, optimized water adsorption energy, and the fast transformation efficiency of adsorbed H into H_2 (Fig. 16d-f). However, these CuFeS_2 nanosheets also have great potential for electrocatalyzing the OER.

In brief, optimizing the density of crystallographic facets with a high fraction of low-coordinated atoms or unique asymmetric zigzag structures could be an effective way to improve the electrocatalytic activities of non-layered TMS. However, facet engineering strongly depends on the synthetic methods used for the controllable exposure of highly active TMS facets.

Edge engineering is a universal route for remarkably improving the electrocatalytic activity of non-layered TMS *via* creating more active sites or increasing their number; it is always presented as a change in structure or morphology. Regular arrangements and the design of unique structures of active materials are conducive to the full exposure of active sites on edges. Zhang *et al.* fabricated 3D hierarchical $\text{MoS}_2/\text{Ni}_3\text{S}_2$ nanorod arrays that were nearly vertically aligned on Ni foam ($\text{MoS}_2/\text{Ni}_3\text{S}_2/\text{NF}$) *via* a one-step hydrothermal procedure.¹⁴⁴ Thanks to the elaborately designed and well-aligned architecture, the $\text{MoS}_2/\text{Ni}_3\text{S}_2/\text{NF}$ electrode with a large surface area and abundant active sites owned excellent electrocatalytic activity and durability for water electrolysis in an alkaline electrolyte. The hierarchical nearly vertically aligned structure possessed enough space for the diffusion of electrolyte into the well-exposed active sites, thus notably increasing the effective electrochemically active surface area of the electrode.

Guan *et al.* prepared self-supported hollow CoS_2 nanotube arrays with hierarchical pores and rich active sites on a flexible substrate, which were derived from uniform wire-like metal-organic framework (MOF) nanoarrays.¹⁴⁵ The open pore space and numerous nanoscale grains of hollow CoS_2 nanotube arrays could provide a large surface area, a high density of active sites, and enhanced gas releasing abilities. The well-defined connection of active CoS_2 with the flexible support ensured high levels of electron transport and good electrode mechanical stability. Consequently, the unique hollow CoS_2 nanotube arrays exhibited high electrochemical activity, with a cell voltage of 1.67 V required to deliver a current density of 10 mA cm^{-2} .

It is worth mentioning that the nanocrystallization of TMS materials is also a simple and promising way to increase the number of active sites *via* edge engineering. Yan *et al.* synthesized mulberry-like NiS/Ni nanoparticles *via* a surface modification strategy.¹⁴⁶ Unique nanostructures of NiS/Ni could provide more active edges sites. Therefore, NiS/Ni nanoparticles exhibited excellent electrocatalytic activity in alkaline solution with remarkably low Tafel slopes of 46 and 74 mV dec^{-1} and small overpotentials of 301 and 161 mV for the OER and HER, respectively. In fact, all the TMS nanostructures mentioned here demonstrated the importance of nanocrystallization.

Strain engineering offers a novel path for promoting the electrocatalytic performance of non-layered TMS for efficient water splitting *via* inducing more active sites.¹⁴⁷ The effects of strain engineering on layered NbS_2 and MoS_2 monolayers have been proved by Chen *et al.*¹⁴⁸ They considered a wide strain range covering both tensile (0-10%) and compressive (0-6%) regions. Biaxial tensile strain enhanced the HER activity of layered TMS monolayers more effectively than uniaxial tensile strain, while compressive strain deteriorated the HER activity. Theoretical calculations revealed that tensile strain could reduce the adiabatic proton affinity but simultaneously increase the adiabatic electron affinity to a larger extent, activate relatively inert inner valence electrons, and further enlarge the d-band exchange splitting, all of which could enhance the electrocatalytic HER activity of layered TMS.

Conceivably, this strategy is also suitable for non-layered TMS. Shan *et al.* predicted strain-induced changes in the

electronic structure and electrocatalytic performance of a 2D Fe_3S_4 material.¹⁴⁹ The calculation results revealed that the half-metallic features of Fe_3S_4 would be converted into metallic features due to the small lattice deformation in Fe_3S_4 monolayers induced by applied external strain, which could optimize the Gibbs adsorption free energy of hydrogen and improve the electron transfer capabilities (Fig. 17a–c). However, more efforts should be carried out to investigate the influence of strain engineering on the OER performance of TMS-based electrocatalysts.

The defect engineering of non-layered TMS is not naturally independent, and it is usually combined with the formation of heterogeneous interfaces,¹⁵¹ unique architecture design,¹⁵² and heteroatom doping.¹⁵³ Defects always have a positive effect on the adsorption activation energy of reactants and the dissociation energy of products in overall water splitting, facilitating the generation and conversion of intermediates *via* regulating the electronic structure.¹⁵⁴ Here, attention is focused on vacancy defects in the lattice structure, which can be induced *via* special techniques such as plasma etching.⁸⁵ Theoretical calculations based on first-principles studies revealed the effects of single cobalt and sulfur vacancies on structural and electronic changes *via* making pyrite CoS_2 samples with high densities of defects and inducing localized defect states in the gap range of the minority spin channel.¹⁵⁵ Yang *et al.* developed highly porous Co–S films that were grown *in situ* on Ni foam, which had superior electrocatalytic activity and a robust nature for water electrolysis.¹⁵⁶ Self-supported Co–S film with a S content level of 21.7 at% could drive a high current density of 500 mA cm^{-2} with low overpotentials of 368 and 155 mV for the OER and HER, respectively. The optimal S content in Co–S film could create more vacancy defects, which would give rise to the proliferation of electrocatalytically active sites or oxygen deficiencies during the OER, significantly enhancing the intrinsic activity and reaction kinetics for both the HER and OER. As reported, sulfur vacancies and strain could be used to finely tune the HER electrocatalytic activity of monolayer MoS_2 .⁴¹ Similarly, Zhang *et al.* proposed that sulfur vacancies and nitrogen dopants in CoS_2 could synergistically activate and optimize the HER activity *via* enhancing the activities of neighboring Co sites with optimal hydrogen adsorption free energies, as shown in Fig. 17d–f.¹⁵⁰

All in all, non-layered TMS are the most extensively studied bifunctional electrocatalysts for water electrolysis with the advantages of being highly active, chemically stable, widely available, and cheap. To further improve their bifunctional electrocatalytic performances, many strategies have been proposed for increasing the number of active sites and enhancing the intrinsic activity and electrical conductivity, such as composite design, heteroatom doping, strain engineering, facet engineering, edge engineering, and defect engineering. To now, studies of non-layered TMS bifunctional electrocatalysts have mainly focused on Ni-based and Co-based sulfides. A summary of non-layered TMS-based bifunctional electrocatalysts is shown in Table 2. N- CoS_2 /Ni foam requires a quite low overpotential of 28 mV to reach 10 mA cm^{-2} for the HER in 1.0 M KOH, exhibiting the best electrocatalytic HER activity.¹³³

Moreover, to reach a current density of 10 mA cm^{-2} in the OER, carbon dots/ NiCo_2S_4 / Ni_3S_2 composites require the smallest overpotential of 116 mV in 1.0 M KOH.¹⁵⁷ Surprisingly, these non-layered TMS bifunctional electrocatalysts possess the potential to surpass noble-metal oxides in overall water splitting, showing far-reaching significance for large-scale water electrolysis applications.

5. Conclusions and outlook

In summary, we have outlined the research progress relating to bifunctional electrocatalysts based on transition metal sulfides for effective overall water splitting, showing their potential as candidates to replace noble-metal-based electrocatalysts. The fabrication of TMS electrocatalysts *via* ‘bottom-up’ and ‘top-down’ strategies has been summarized. Typically, exfoliation and ball milling approaches are the leading ‘top-down’ strategy processing methods, which are suitable for synthesizing nano-structured and micro-structured TMS on a large scale from bulk TMS materials. Additionally, hydrothermal, solvothermal, chemical vapor deposition, and solid-phase synthesis approaches are the main representatives of the ‘bottom-up’ strategy, showing great potential for the preparation of self-supported TMS electrodes when suitable reaction substrates are adopted. *Via* these controlled fabrication approaches, the rational tuning of well-defined TMS shapes, sizes, hierarchical structures, and defect states could be achieved, with great influence on the electronic structure and electrocatalytic performance of TMS. Importantly, the active sites of TMS electrocatalysts are nearly always metal atoms, while exposed S atoms and S vacancies also play a vital direct role in the HER *via* increasing the number of active sites and changing the coordination environments of adjacent metal atoms. Moreover, the indirect role of S atoms in the HER provides locations for hydrogen adhesion and separation when metal atoms act as the active sites. However, TMS tend to function as pre-catalysts in complex OER processes, with involvement in conversion to transition metal hydr(oxy)oxides.

Layered TMS (*e.g.*, MoS_2 , WS_2 , and TaS_2) exhibit excellent intrinsic electrocatalytic HER activity due to their well-exposed edge sites and good electrical conductivity, while their direct application in the OER is limited. To further explore the potential of layered TMS for use in water splitting, numerous efforts have been carried out to optimize their electrochemical properties *via* improving their intrinsic electrical conductivity and stability and enhancing their chemisorption capabilities for O-containing species. Phase engineering, composite design, and heteroatom doping are favorable strategies for achieving this goal. Phase engineering is a promising way to enhance the electrocatalytic OER activity of layered TMS, as the 1T phase usually exhibits higher OER activity than the 2H phase. Integrating layered TMS with other materials with highly active OER performance or good conductivity can be used to construct bifunctional electrocatalysts with novel composites/heterostructures utilizing synergistic effects relating to the counterparts. Lastly, heteroatom doping is favorable for modulating the electronic structures of layered TMS and

optimizing the chemical-adsorption energies of reactive intermediates during the water splitting reactions, and for creating more active sites *via* generating more defects and distortions in the lattice. Heteroatom doping can be divided into two types, namely substitutional doping and surface charge transfer doping, depending on the interactions between heteroatoms and layered TMS.

Non-layered TMS are the most extensively studied bifunctional electrocatalysts for overall water splitting, with the advantages of being widely available, inexpensive, chemically stable, and highly active, and examples include FeS₂, Ni₃S₂, NiS₂, CoS₂, and Co₃S₄. Many strategies have been researched to improve their bifunctional electrocatalytic performance *via* increasing the number of active sites and enhancing the intrinsic activity and electrical conductivity, *e.g.*, composite designing, heteroatom doping, strain engineering, facet engineering, edge engineering, and defect engineering. Similar to layered TMS, the composite designing of non-layered TMS can enhance the charge transfer kinetics and create new active sites due to synergetic effects between the components and the triggering of additional interfaces and defects. Heteroatom doping also can be used to regulate electronic structures to acquire faster electron transfer and optimize the binding energies of intermediates with non-layered TMS. The development of ternary and quaternary TMS-based materials with excellent redox reversibility and high electronic conductivity has created effective substitutes for noble-metal-based electrocatalysts. Facet engineering is strongly dependent on the synthetic methods used for the controllable exposure of the highly active facets of non-layered TMS, and it is favorable for improving the intrinsic activity. Strain engineering and edge engineering can change the electronic character and induce the creation of more active sites, respectively, and these methods are worth further exploration. Finally, as a universal method for optimizing the electrocatalytic performance of non-layered TMS, defect engineering is not a naturally independent method, and it is usually combined with the formation of heterogeneous interfaces, unique architecture design, and heteroatom doping. In principle, most strategies for enhancing the performance of non-layered TMS are also suitable for use with layered TMS whose OER activities need to be further exploited.

Although a lot of progress has been made related to the development of TMS-based bifunctional electrocatalysts, some challenges still exist, as follows:

(1) The electrocatalytic reaction mechanisms related to these bifunctional electrocatalysts in overall water splitting have still not been fully revealed, and more theoretical study efforts should be devoted to uncovering them and providing strong support for exploiting advanced electrocatalysts. On one hand, the detailed reaction mechanism of the OER requires more investigation to uncover its true intrinsic nature. As a sluggish four electron–proton coupled reaction, the several OH*, O*, and OOH* intermediates formed during the OER make it a great challenge to evolve a consolidated reaction mechanism, and different reaction mechanisms for the OER have been proposed by different research groups.²⁴ On the other hand, the HER

reaction mechanism at the atomic level is still ambiguous at present, and it requires more in-depth investigation. As is known, the HER mechanism in alkaline electrolytes is totally different from that in acidic media, where it only involves hydrogen binding energetics. The HER mechanism in alkaline media is more complex, including the adsorption/dissociation of water molecules, the adsorption/desorption of hydrogen, and even the OH[−] poisoning of the electrocatalyst surface. To further explore the reaction mechanisms, advanced *in situ* characterization techniques, such as *in situ* X-ray absorption spectroscopy (XAS), *in situ* Raman spectroscopy, and surface interrogation scanning electrochemical microscopy (SI-SECM), should be widely developed and applied to explore dynamic changes in electronic structures, the dynamic evolution of intermediates, and the binding strengths of adsorbates on the surfaces of electrocatalysts. Nowadays, theoretical calculations based on the first-principles calculations provide powerful support when exploring reaction mechanisms at an atomic level. More importantly, multi-scale modelling approaches involving atomic-level models (such as DFT) and energy minimization techniques (molecular dynamics and Monte Carlo models) could be promising for understanding electrocatalytic processes. Combining theoretical calculations with advanced experimental characterization techniques could provide more in-depth understanding of HER and OER mechanisms.

(2) The development of new or combined approaches for further improving the activity of TMS-based electrocatalysts is essential. The direction of efforts to improve the electrocatalytic activities of TMS-based materials is aimed at increasing the number of active sites, improving the electrical conductivity, and optimizing the absorption energies of reaction intermediates. In this review, several strategies for enhancing the HER and OER performances of TMS-based electrocatalysts have been summarized. The approaches of phase engineering, heteroatom doping, strain engineering, and composite designing are favorable for improving the electron transfer characteristics of materials, and the approaches of edge engineering, facet engineering, and defect engineering are beneficial for increasing the number of exposed active sites. The effects of strain engineering, especially on the OER activities of TMS-based materials, need to be further investigated from a mechanistic perspective and examined through experiments. Utilizing multiple approaches simultaneously to realize a maximized enhancement effect is feasible and effective, for example integrating heteroatom doping and strain engineering together⁴¹ or integrating heteroatom doping and composite design together.¹⁵⁸ In this regard, the synergistic effects of different optimization methods provide a fascinating way to enhance the electrocatalytic activities of TMS-based materials. Examples include preparing highly active heteroatom-doped TMS-based composites with low-dimensional nanostructures *via* coupling composite designing, edge engineering, and heteroatom doping, and designing metallic-phase TMS-based composites with special morphologies *via* coupling composite designing, phase engineering, and edge engineering.

(3) Although enhancements in the activities of TMS-based electrocatalysts are fruitful and activities can even be

comparable with noble-metal-based electrocatalysts, the stabilities of most bifunctional electrocatalysts cannot meet the needs of industrial applications completely, as they suffer from susceptibility to corrosion and the destruction of structures during long-term water electrolysis. Moreover, the large-scale preparation of TMS-based electrodes is also a bottleneck for industrial applications. Hence, robust materials and electrodes that can be manufactured on a large scale need to be further investigated urgently. On one hand, the creation of TMS-based materials decorated with protective or sacrificial materials could be a simple method to improve corrosion and oxidation resistance. On the other hand, the *in situ* growth of active TMS on conductive substrates that could be employed as self-supported electrodes is quite a promising approach for dramatically enhancing durability. Strong interactions between active TMS and substrates could dramatically enhance the electrocatalytic durability and electron transfer during long-term service. *In situ* growth can also avoid the usage of carbon additives and polymer binders, providing more exposed active sites and reducing cost. Broadly speaking, when employing conductive substrates during the preparation process, the 'bottom-up' approaches of hydrothermal, solvothermal, chemical vapor deposition, and solid-phase synthesis all exhibit potential for synthesizing self-supported electrodes on a large scale. In this case, the development of reaction devices capable of the large-scale preparation of TMS-based materials is critical for satisfying industrial application demands. Additionally, advanced preparation methods suitable for the large-scale fabrication of these novel electrocatalysts require more attention and development, such as atomic layer deposition, spray pyrolysis, and pulsed laser deposition.

(4) Currently, reported bifunctional electrocatalysts are mostly employed in alkaline electrolytes, while only a few have been applied in acidic and/or neutral media, as the instability of TMS-based electrocatalysts seriously limits their applications.¹⁵⁹ However, universal-pH bifunctional electrocatalysts with superior activity and durability in near-neutral media are still required, as these would have great significance for the industrial large-scale development of seawater electrolysis. For example, other binary, ternary, or multiple sulfides with precisely defined shapes, sizes, and compositions, including Sb, Sn, Mn, Re, Cu, and Zn metal based examples, deserve more attention. In addition, bifunctional electrocatalysts for overall urea-water electrolysis are highly desired,¹⁶⁰ as they can be applied to efficient H₂ generation and the purification of urea-rich wastewater to meet the requirements of both energy conversion and environmental protection.

All in all, developing bifunctional TMS-based electrocatalysts for overall water splitting is very appealing yet challenging. With continuing research efforts being focused on the strategies mentioned above, it is conceivable that TMS-based bifunctional electrocatalysts with extremely improved activity and long-term stability will be developed for industrial applications.

Conflicts of interest

There are no conflicts to declare.

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