

## PAPER

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# A voltage-controllable VO<sub>2</sub> based metamaterial perfect absorber for CO<sub>2</sub> gas sensing application

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Vanadium dioxide (VO<sub>2</sub>) based metamaterial perfect absorbers (MPAs) have high potential application values in sensing gas molecules. However, a tuning mechanism *via* temperature manipulation lacks the compatibility with electronic devices. In this study, a voltage-controllable device is proposed by integrating an MPA and micro-electro-mechanical system (MEMS) based microheater for CO<sub>2</sub> gas sensing application. The MPA is composed of a metal–dielectric–metal (MDM) structure and tailored to form an H-shaped metamaterial. The central bar of the H-shaped metamaterial is composed of a VO<sub>2</sub> material, which exhibits perfect absorption in the CO<sub>2</sub> gas absorption spectrum, *i.e.*, at a wavelength of 2.70 μm. The intergated microheater is patterned by using fractal theory to provide high heating temperature and high uniformity of surface temperature. By precisely driving a DC bias voltage on the microheater, the MPA is heated and it can exhibit switchable optical properties with high efficiency. These results provide a strategy to open an avenue for sensors, absorbers, switches, and programmable devices in infrared wavelength range applications.

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## 1. Introduction

Recently, metamaterial absorbers have shown good prospects due to their ability to control the propagation of electromagnetic (EM) waves.<sup>1–4</sup> They provide the opportunities to realize unique EM properties for widespread applications, such as invisibility cloaks, perfect absorbers, super lenses, and solar cells.<sup>5–10</sup> Metamaterial perfect absorbers (MPAs) have been proposed to realize nearly 100% absorption in the wide EM spectrum from microwave,<sup>11</sup> and terahertz (THz)<sup>12</sup> to infrared (IR)<sup>13–15</sup> wavelength ranges. Since many surrounding gas molecules exhibit identical vibration frequency to IR light, they can be distinguished in the IR wavelength range. The development of an MPA-based IR gas sensor has become a research hotspot owing to its high selectivity and high robustness.<sup>16</sup> Many MPA-based sensors have been demonstrated with powerful performances.<sup>17–19</sup> However, they cannot be actively tuned once fabricated while the tunable characteristic is desired in real applications by using an external stimulus. Micro-electro-mechanical system (MEMS) technology is one of the most important tuning mechanisms to control the MPA.<sup>20</sup> By directly adjusting the geometrical dimensions of the metamaterial unit cell, the MEMS-based metamaterial exhibits a large tuning range and good linearity in sensing applications.<sup>21–24</sup>

Furthermore, vanadium dioxide (VO<sub>2</sub>) based metamaterials are widely studied for their active tunability *via* temperature manipulation.<sup>25</sup> VO<sub>2</sub> is a well-known material while the insulator-to-metal transition characteristic can be induced thermally, optically or electrically at the time scale of picoseconds due to the atomic level deformation for sensing applications or active phase change material-based control.<sup>26–32</sup> By improving the internal temperature above the critical temperature, dramatic changes of electrical conductivity can be realized in VO<sub>2</sub>. They are reflected in the optical dielectric properties of VO<sub>2</sub> to control the propagation of EM waves with high efficiency. Due to the ultrafast switching time of VO<sub>2</sub>, it is demonstrated to be integrated with metamaterial for numerous practical applications, such as modulators, programmable devices, and tunable antennas.<sup>33–36</sup>

In this work, a CO<sub>2</sub> gas sensor is proposed by integrating an H-shaped MPA and MEMS-based microheater to form a voltage-controllable device. The microheater is patterned using mathematical fractal theory and named the Peano-shaped microheater. The thermal emission of the Peano-shaped microheater can fully cover the heating area uniformly. There are many literature reported VO<sub>2</sub> based metamaterials to investigate their tunable EM properties by changing heating temperature.<sup>37</sup> While such a thermal tuning method lacks the compatibility with electronic devices, an integrated MEMS-based microheater is desired to provide a precise voltage-controllable characteristic. The Peano-shaped microheater can collect all flowing electric energy and then convert it into thermal energy. By increasing the heating temperature of the Peano-

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shaped microheater, the surface temperature is distributed uniformly compared with that of a traditional winding-shaped microheater. The MEMS-based MPA can exhibit perfect absorption with precisely controlled voltage at a wavelength of 2.70  $\mu\text{m}$ , which is the  $\text{CO}_2$  absorption spectrum. It is suitable to be used as a  $\text{CO}_2$  gas sensor with high sensitivity. Moreover, such perfect absorption can be continuously switched to be applied in many fields, such as IR switches, programmable IR devices, and IR detectors.

## 2. Designs and methods

Fig. 1 illustrates the 3D schematic drawing and tuning mechanism of MEMS-based MPA design. Such a design is composed of an H-shaped MPA on the top and a Peano-shaped microheater on the bottom. The compositions of the H-shaped MPA are Au/ $\text{VO}_2$  on  $\text{SiO}_2$  and Au from top to bottom. The top Au and  $\text{VO}_2$  thin-films are 300 nm in thickness, and are tailored to form an H-shaped periodic array. The middle transverse bar is patterned by using the  $\text{VO}_2$  material to provide active tunability as shown in the inserted image of Fig. 1. Meanwhile, the key parameters of the H-shaped MPA unit cell are also denoted in Fig. 1, including the thickness of the dielectric layer ( $t$ ), and lengths of the longitudinal bar ( $l_1$ ), and transverse bar ( $l_2$ ). The period ( $P$ ) of the H-shaped MPA unit-cell is kept constant at 2.3  $\mu\text{m}$ . For the Peano-shaped microheater, an Au thin-film is deposited and patterned by using mathematical fractal theory on the Si substrate. The high resistance heat is generated by the flowing electric current in the metallic loop. Such a Peano-shaped microheater provides high heating temperature and high uniformity of surface temperature. The heating area is  $200 \times 200 \mu\text{m}^2$ . The dielectric layer between the top H-shaped MPA and the bottom Peano-shaped microheater is composed of  $\text{SiO}_2$  material, whose thickness is 5 nm to perform electrical insulating and thermal conducting functions. By driving a

direct current (DC) bias voltage to the electrodes of the microheater, some heat is generated and it flows to the top metamaterial pattern, which can improve the internal temperature of the metamaterial with high efficiency. At room temperature, the  $\text{VO}_2$  thin-film is operated in the insulating state. By increasing the heating temperature higher than the phase transition temperature, the structural transformation is induced and the  $\text{VO}_2$  bar of the MPA turns into a metallic phase. The complex dielectric properties of the  $\text{VO}_2$  thin-film in both metallic and insulating phases can be described by a classical Drude–Lorentz model, which is expressed by<sup>38</sup>

$$\varepsilon(\omega) = \varepsilon_\infty - \frac{\omega_n^2}{\omega^2 + i\omega_c\omega} + \sum_{i=1}^n \frac{s_i}{1 - \omega^2/\omega_i^2 - i\Gamma_i\omega/\omega_i} \quad (1)$$

where  $\varepsilon_\infty$  is a constant contribution to the real part of the dielectric constant from high-frequency electronic transition,  $\omega_n$  is the carrier density parameter,  $\omega_c$  is the collision frequency,  $s_i$  is the strength, and  $\Gamma_i$  is the linewidth. In the third term of eqn (1), the strength  $s_i$ , resonant frequency  $\omega_i$  and scattering rate  $\Gamma_i/\omega_i$  are temperature dependent parameters due to the interband transitions of bound electrons effected by temperature. The EM characteristics of the proposed H-shaped MPA are achieved through Lumerical solution's finite difference time domain-based simulations. The perfectly matched layer boundary conditions are assumed in the  $z$ -axis direction to evaluate the absorptivity, while the periodic boundary conditions are adopted in both the  $x$ - and  $y$ -axis directions to calculate the coupling effect between the neighboring metamaterials.

## 3. Results and discussion

The propagation of EM waves in the H-shaped MPA is controllable by appropriately manipulating the geometric dimensions. According to the effective medium approximation, the

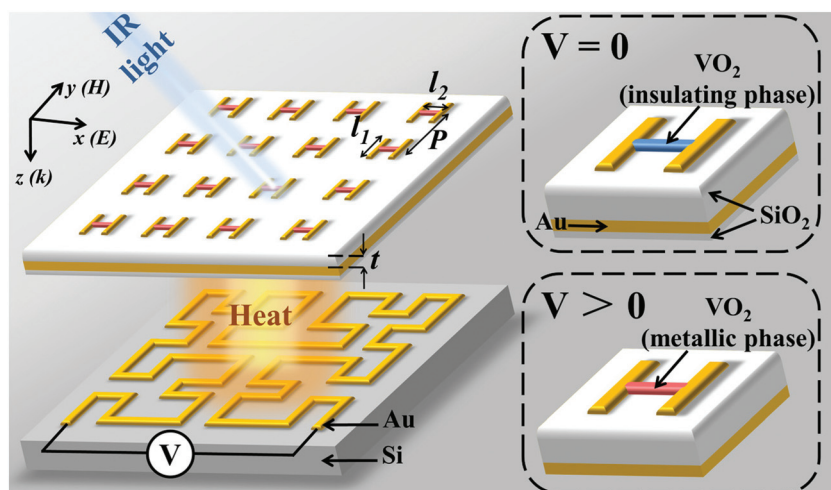


Fig. 1 Schematic drawing and tuning mechanism of MEMS-based MPA design. The periods of unit cells are kept as  $P = 2.3 \mu\text{m}$ .

effective refractive index and impedance index of metamaterials are calculated by<sup>39</sup>

$$n_{\text{eff}} = \frac{1}{kd} \cos^{-1} \left( \frac{1 - r^2 + t^2}{2t} \right) \quad (2)$$

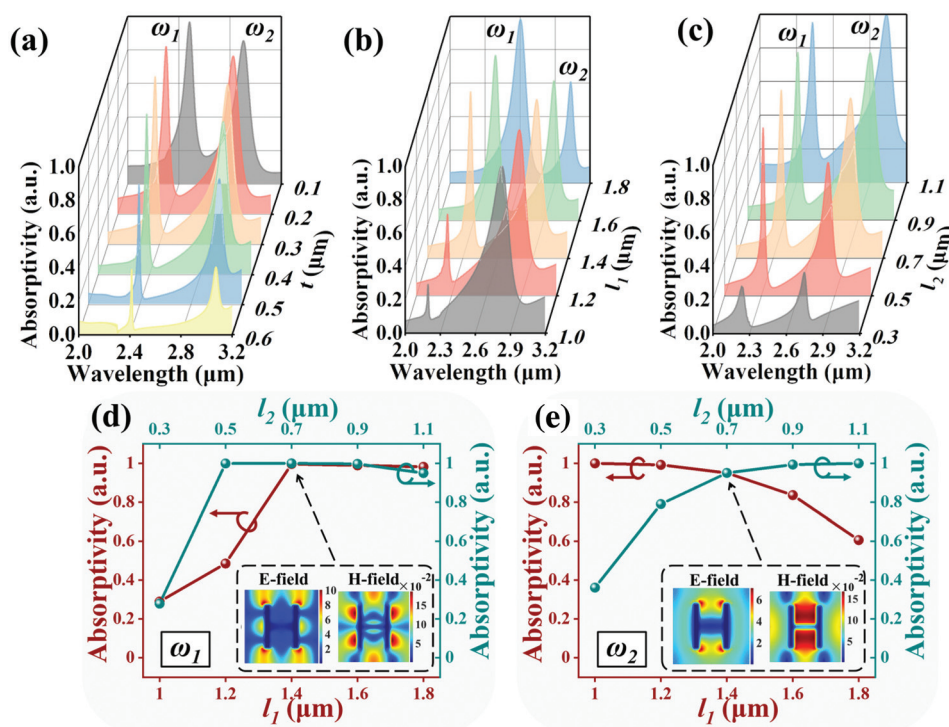
$$z_{\text{eff}} = \sqrt{\frac{(1+r)^2 - t^2}{(1-r)^2 - t^2}} \quad (3)$$

$$\varepsilon_{\text{eff}} = n_{\text{eff}} / z_{\text{eff}} \quad (4)$$

$$\mu_{\text{eff}} = n_{\text{eff}} z_{\text{eff}} \quad (5)$$

where  $\varepsilon_{\text{eff}}$  and  $\mu_{\text{eff}}$  are the effective permittivity and permeability of the metamaterial, respectively,  $r$  is the reflection coefficient,  $t$  is the transmission coefficient,  $k$  is the incident wave vector, and  $d$  is the metamaterial thickness. By appropriately designing the dielectric properties of VO<sub>2</sub>, the effective impedance of the H-shaped MPA can be matched with that of the free space and then achieve zero reflection and perfect absorption. Three key parameters of the H-shaped MPA are investigated to optimize the optical properties, including  $t$ ,  $l_1$  and  $l_2$  values. Herein, they are individually adjusted while the other two parameters are kept constant. Meanwhile, the VO<sub>2</sub> bar is assumed to be phase transitioned in the metallic state. In the absorption spectra of Fig. 2, it can be clearly observed that such designs exhibit a dual-band characteristic in the IR wavelength range from 2.00  $\mu\text{m}$  to 3.20  $\mu\text{m}$ . The resonance operated at a short wavelength is denoted as  $\omega_1$  while the resonance

operated at a long wavelength is denoted as  $\omega_2$ . The  $t$  parameter determines the absorption intensity by affecting the coupling efficiency between the H-shaped metamaterial and the bottom Au mirror layers as shown in Fig. 2(a). By increasing the  $t$  value from 0.1  $\mu\text{m}$  to 0.6  $\mu\text{m}$ , the  $\omega_1$  resonance of the H-shaped MPA is blue-shifted from the wavelength of 2.49  $\mu\text{m}$  to 2.41  $\mu\text{m}$ , while the absorptivity is enhanced and then decays from 0.97 to 0.39. Similarly, the  $\omega_2$  resonance is red-shifted from the wavelength of 2.91  $\mu\text{m}$  to 3.07  $\mu\text{m}$ , while the absorptivity is enhanced and then decays from 0.85 to 0.40. When the  $t$  value is 0.3  $\mu\text{m}$ , the  $\omega_1$  resonance achieves perfect absorption. Therefore, the condition of 0.3  $\mu\text{m}$  is suitable for the design of the H-shaped MPA with perfect IR absorption application. In Fig. 2(b and c), the variations of  $l_1$  and  $l_2$  values also affect the coupling efficiency of the H-shaped MPA by controlling the dual-resonance in the  $x$ - $y$  plane. By increasing the  $l_1$  value, the  $\omega_1$  resonance of the H-shaped MPA is red-shifted from the wavelength of 2.20  $\mu\text{m}$  to 2.61  $\mu\text{m}$ , while the  $\omega_2$  resonance is red-shifted from the wavelength of 2.83  $\mu\text{m}$  to 3.04  $\mu\text{m}$ . Herein, the absorptivity is essential in the sensing application and the relationships of absorptivity of the H-shaped MPA with different  $l_1$  and  $l_2$  values at  $\omega_1$  and  $\omega_2$  resonances are summarized in Fig. 2(d) and (e), respectively. By increasing the  $l_1$  value from 1.0  $\mu\text{m}$  to 1.8  $\mu\text{m}$ , the absorptivity of the  $\omega_1$  resonance is enhanced from 0.29 to 0.98 and that of the  $\omega_2$  resonance decays from 0.99 to 0.61. In addition, the  $l_2$  parameter also determines the EM properties of the H-shaped MPA. By increasing the  $l_2$  value from 0.3  $\mu\text{m}$  to 1.1  $\mu\text{m}$ , the  $\omega_1$  reso-

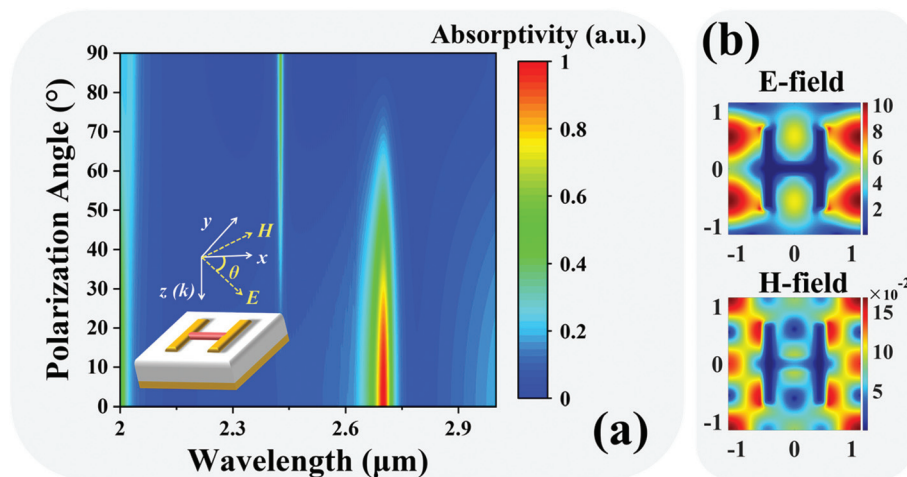


**Fig. 2** Absorptivity of the H-shaped MPA with different (a)  $t$ , (b)  $l_1$ , and (c)  $l_2$  values, respectively. The relationships of the resonance intensities of the H-shaped MPA with different  $l_1$  and  $l_2$  values at (d)  $\omega_1$  and (e)  $\omega_2$  resonances, respectively. The inserted images show the corresponding  $E$ -field and  $H$ -field distributions under the conditions of  $l_1 = 1.4 \mu\text{m}$  and  $l_2 = 0.7 \mu\text{m}$ .

nance is red-shifted from the wavelength of  $2.24\ \mu\text{m}$  to  $2.43\ \mu\text{m}$  and the  $\omega_2$  resonance is red-shifted from the wavelength of  $2.75\ \mu\text{m}$  to  $3.05\ \mu\text{m}$ . The absorptivity of the  $\omega_1$  resonance is enhanced from 0.28 to 0.95 and that of the  $\omega_2$  resonance is enhanced from 0.36 to 0.99. Such strong resonances can be explained by the corresponding  $E$ - and  $H$ -field distributions as shown in the inset images of Fig. 2(d) and (e), which show strong  $E$ -field energies around the periodic unit cells of the H-shaped MPA. In view of the above results, the optimized parameters of the H-shaped MPA are  $t = 0.3\ \mu\text{m}$ ,  $l_1 = 2.0\ \mu\text{m}$ , and  $l_2 = 0.7\ \mu\text{m}$  to achieve perfect absorption at a wavelength of  $2.70\ \mu\text{m}$  for  $\text{CO}_2$  sensing application. Furthermore, the absorptivity of the H-shaped MPA can be tuned by changing the geometric dimensions for IR variable optical attenuator (VOA) application.

Since the *meta*-atom of the H-shaped MPA shows an anisotropic pattern, the EM properties can be controlled by incident polarization light. The absorption spectra operated at different polarization angles ( $\theta$ ) are illustrated in Fig. 3(a). To clearly evaluate the polarization-dependent characteristic of MPA,  $\theta$  is defined as the included angle of the  $E$ -field and  $x$ -axis direction. Herein,  $\theta$  equals  $0^\circ$  is regarded as the transverse electric (TE) mode and  $90^\circ$  is regarded as the transverse magnetic (TM) mode. It is clearly observed that the dual-band characteristic of the H-shaped MPA is gradually tuned to a single-band characteristic by changing the  $\theta$  value from  $0^\circ$  to  $90^\circ$ . It can serve as a polarizer in the IR wavelength range. Under the condition of  $\theta = 90^\circ$ , the H-shaped MPA exhibits the maximum absorptivity of 0.44 at a wavelength of  $2.40\ \mu\text{m}$ . The corresponding  $E$ - and  $H$ -field distributions are shown in Fig. 3(b). The  $H$ -field distributions in the TM mode at a wavelength of  $2.40\ \mu\text{m}$  are much weaker than those in the TE mode at a wavelength of  $2.70\ \mu\text{m}$ . It explains the lower absorption which is caused by the TM mode. These polarization-dependent characteristics of the H-shaped MPA can be potentially used in IR polarization switching, VOA, and sensing applications.

To endow the H-shaped MPA with active tunability, the Peano-shaped microheater is integrated to control the internal temperature of the H-shaped MPA with high efficiency. This voltage-controllable metamaterial design is desired to realize large-scope applications and it is worth investigating the electrical and thermal properties of the device. The relationships between the applied DC voltages and heating temperatures for the Peano-shaped microheater and the conventional winding-shaped microheater are illustrated in Fig. 4(a). The material of the microheater is adopted as the Au material with a thermal conductivity of  $317\ \text{W}(\text{m K})^{-1}$ . By driving a DC bias voltage on the electrodes of microheaters, the resistance heat can be generated and conducted to the H-shaped MPA to improve the surface temperature. In real applications, the top metamaterial layer is designed on the central area of the microheater while the area of the metamaterial is much smaller than that of the microheater. Therefore, the heating temperature is adopted as the maximum surface temperature. Moreover, the thickness of the  $\text{SiO}_2$  layer between the microheater and MPA is  $5\ \text{nm}$  to perform the excellent thermal conducting function. In such a design, the thermal energy is highly efficiently and highly uniformly propagated to the metamaterial layer. It can be clearly observed that the Peano-shaped microheater shows a higher temperature of  $379\ \text{K}$  and the winding-shaped microheater is  $370\ \text{K}$  at a driving voltage of  $2.0\ \text{volts}$ . It means that the Peano-shaped microheater exhibits less thermal dissipation with identical electrical power and then the energy consumptions can be further reduced. The heating power is concentrated in the central area of the microheaters. The effective heating area covers over 75% and exhibits the maximum temperature over 90%. It endows the H-shaped MPA with high-robust EM properties. Thus, the insulator-to-metal transition of  $\text{VO}_2$  can be induced thermally, which will increase the conductivity of  $\text{VO}_2$  and reduce the optical transmission in the near-IR wavelength range. These results reflect the variation of optical dielectric properties. Fig. 4(b) shows the relationship between the index



**Fig. 3** (a) Absorptivity of the H-shaped MPA at different polarization angles. (b) The corresponding  $E$ -field and  $H$ -field distributions with a polarization angle of  $90^\circ$  at a wavelength of  $2.40\ \mu\text{m}$ .



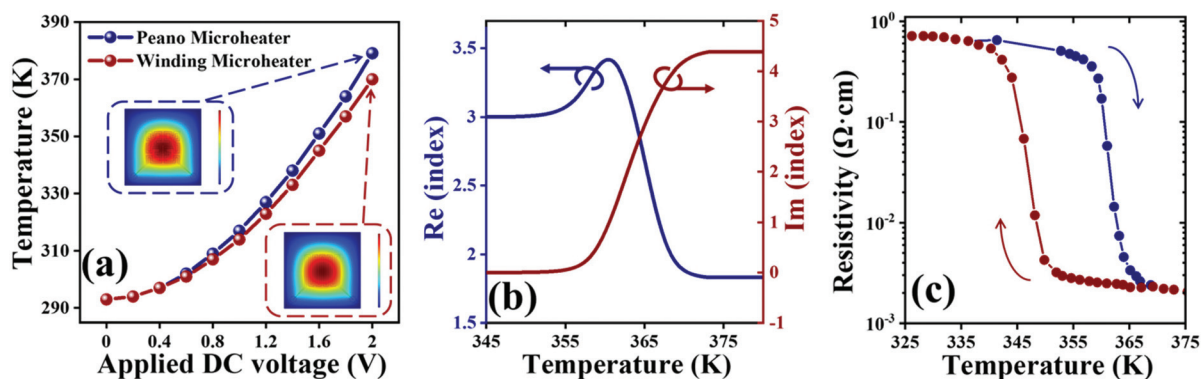


Fig. 4 (a) The relationships between the applied DC voltages and heating temperatures of the Peano-shaped microheater. The inserted images show the temperature distributions of the Peano-shaped microheater with a driving voltage of 2.0 volts. (b) Variation of the index of  $\text{VO}_2$  with different heating temperatures at a wavelength of 2.70  $\mu\text{m}$ . (c) Variation of the resistivity of  $\text{VO}_2$  with different heating temperatures.

of  $\text{VO}_2$  and heating temperature. Since the relationship is dependent on the operating wavelength and the MEMS-based MPA is proposed for  $\text{CO}_2$  sensing application, the variations of the index are calculated using the  $\text{CO}_2$  absorption spectrum, *i.e.*, at a wavelength of 2.70  $\mu\text{m}$ . By increasing the heating temperature, the real part of the index gradually increases from 3.00 to 3.42 and then sharply decays to 1.83 at the stable metallic state. Meanwhile, the imaginary part of the index increases from 0.01 to 4.38. The relationships between the heating temperatures and resistivity for  $\text{VO}_2$  are illustrated in Fig. 4(c).<sup>38,40</sup> By increasing the heating temperature, the resistivity can be reduced from 0.689  $\Omega \cdot \text{cm}$  to 0.002  $\Omega \cdot \text{cm}$  due to the insulator-to-metal transition characteristic of  $\text{VO}_2$ . These results prove that the Peano-shaped microheater is suitable to be used to control the optical dielectric properties of the  $\text{VO}_2$  material.

The insulator-to-metal transition of the  $\text{VO}_2$  material provides the opportunity to control the absorptivity of the MEMS-based MPA by changing the heating temperature. The absorption spectra of the MEMS-based MPA with different applied DC voltages in the TE mode are illustrated in Fig. 5(a). By driving a DC bias voltage of 1.60 volts, the temperature of the MEMS-based MPA is 351 K and the  $\text{VO}_2$  bars are operated at the insulating state. The absorptivity of  $\omega_1$  resonance is 0.25 while the absorptivity of  $\omega_2$  resonance is 0.15. By increasing the applied DC voltage to 1.92 volts, the temperature of the MEMS-based MPA increases to 373 K and the  $\text{VO}_2$  bars are transitioned to the metallic state. The absorption intensities of  $\omega_1$  and  $\omega_2$  resonances are enhanced to 0.99 and 0.95, respectively. Fig. 5(c) shows the non-linear relationship of the applied DC voltages and absorption intensities in the  $\text{CO}_2$  absorption spectrum (wavelength of 2.70  $\mu\text{m}$ ). Although many research studies have reported the optical properties of the  $\text{VO}_2$ -based metamaterial at insulating and metallic states, the continuous tuning mechanism of EM characteristics has been rarely reported. The temperature of the MPA can be precisely controlled by driving a DC bias voltage on the Peano-shaped microheater, and the continuous tuning of the absorption

intensity can be achieved. In addition, Fig. 5(b) shows the absorptivity of the MEMS-based MPA with different applied DC voltages in the TM mode. The absorption intensity is zero at a wavelength of 2.70  $\mu\text{m}$ , which means that the absorption ability of the proposed MEMS-based MPA to  $\text{CO}_2$  inherent resonance is mainly contributed by the incident TE resonances. The proposed MEMS-based MPA shows the perfect absorption characteristic to detect  $\text{CO}_2$  gas in the TE mode. It is suitable to be used as a high-efficiency  $\text{CO}_2$  gas sensor. Fig. 5(d) shows the schematic drawing of the MEMS-based MPA for the  $\text{CO}_2$  gas molecule sensing function. The MEMS-based MPA device can be encapsulated in a microfluidic channel and injected with  $\text{CO}_2$  gas. Therefore, the incident IR light is absorbed by  $\text{CO}_2$  gas molecules. The proposed MPA exhibits absorption characteristic in the  $\text{CO}_2$  absorption spectrum, *i.e.*, at a wavelength of 2.70  $\mu\text{m}$ . When the concentration of  $\text{CO}_2$  gas is low in the surrounding environment, the maximum energy of incident IR light is absorbed by the MPA. When the concentration of  $\text{CO}_2$  gas is high, the incident IR light is decayed and the absorption intensity of the MPA becomes weak. Such absorbed optical energy is generated by the resonance of the MPA and then converted into thermal energy, which can be detected by an external thermopile. In addition, by applying different DC voltages on the microheater, the absorption capability and the detected signal at the operated wavelength can be switched. The difference in the switchable signals can be calculated and exploited to improve the efficiency of the  $\text{CO}_2$  sensor while the noise at the other wavelengths is filtered. While the thermal tuning method lacks the compatibility with electronic devices, the device is designed to provide precise voltage-controllable characteristics. Since the existing  $\text{CO}_2$  sensors are limited by the large size and cannot serve the requirement of high integration, the proposed designs can undergo miniaturization and can be integrated into electronic devices to realize high-efficient  $\text{CO}_2$  sensing application. Moreover, the MEMS-based MPA can be designed to operate at different IR wavelength ranges to detect other gas molecules as shown in Fig. 2.

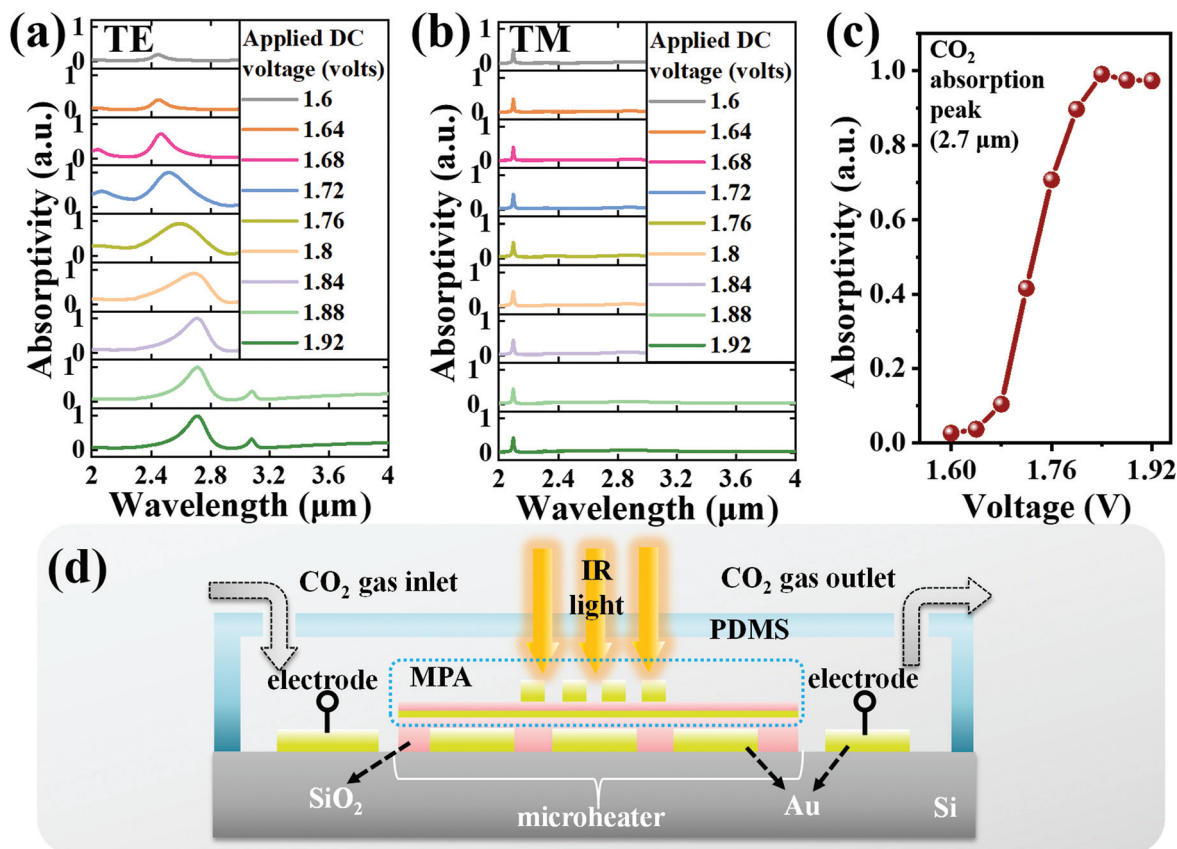


Fig. 5 Absorptivity of the MEMS-based MPA with different applied DC voltages in (a) TE and (b) TM modes. (c) The relationship of applied DC voltages and absorption intensities at a wavelength of 2.70  $\mu\text{m}$ . (d) Schematic drawing of the MEMS-based MPA for the  $\text{CO}_2$  gas molecule sensing function.

## 4. Conclusion

In conclusion, an actively tunable metadvice is proposed by integrating an H-shaped MPA and MEMS-based microheater to form the MEMS-based MPA for  $\text{CO}_2$  sensing application. The microheater is patterned by using mathematical fractal theory and named the Peano-shaped microheater to provide high heating temperature and high uniformity of surface temperature. The heat is generated and then flows to the top H-shaped MPA by driving a DC bias voltage on the Peano-shaped microheater. By exploiting the insulator-to-metal transition characteristic of the  $\text{VO}_2$  material, the  $\text{VO}_2$  based H-shaped MPA can be tuned at a wavelength of 2.70  $\mu\text{m}$  precisely. It is suitable to be used as a  $\text{CO}_2$  gas sensor with high sensitivity. The proposed MEMS-based MPA could be further used as an IR optoelectronic device in widespread applications, such as optical switches, tunable perfect absorbers, VOAs, and polarizers.

## Author contributions

X. X. and R. X. contributed equally to this work. X. X., R. X., and Y.-S. L. conducted the methodology. X. X. and

R. X. conducted the analysis with software and experiment. R. X. drew the schematic diagrams in the paper. X. X. and R. X. led the writing of original draft preparation. X. X. and Y.-S. L. led the review and editing of the paper. All authors discussed the results and commented on the manuscript.

## Conflicts of interest

The authors declare no conflicts of interest.

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## References

- 1 N. I. Landy, S. Sajuyigbe, J. J. Mock, D. R. Smith and W. J. Padilla, *Phys. Rev. Lett.*, 2008, **100**, 207402.
- 2 C. M. Watts, X. L. Liu and W. J. Padilla, *Adv. Mater.*, 2012, **24**, 98–120.
- 3 W. Li, U. Guler, N. Kinsey, G. V. Naik, A. Boltasseva, J. G. Guan, V. M. Shalaez and A. V. Kildishev, *Adv. Mater.*, 2014, **26**, 7959.
- 4 Y. X. Cui, Y. R. He, Y. Jin, F. Ding, L. Yang, Y. Q. Ye, S. M. Zhong, Y. Y. Lin and S. L. He, *Laser Photonics Rev.*, 2014, **8**, 495–520.
- 5 X. Zhang, K. Chen, Y.-S. Lin and B.-R. Yang, *Opt. Mater.*, 2021, **122**, 111679.
- 6 X. Xu, Y.-S. Lin, R. Fang and B.-R. Yang, *Opt. Mater. Express*, 2021, **11**(10), 3577–3586.
- 7 Y.-S. Lin, J. Dai, Z. Zeng and B.-R. Yang, *Nanoscale Res. Lett.*, 2020, **15**, 77.
- 8 Y. Han, J. Lin and Y.-S. Lin, *Opt. Lett.*, 2020, **45**(24), 6619–6622.
- 9 F. Zhan and Y.-S. Lin, *Opt. Lett.*, 2020, **45**(13), 3633–3636.
- 10 D. Zheng and Y.-S. Lin, *Adv. Mater. Technol.*, 2020, **5**, 202000584.
- 11 Z. Q. Du, J. G. Liang, T. Cai, X. Wang, Q. D. Zhang, T. W. Deng, B. R. Wu, R. Q. Mao and D. P. Wang, *Opt. Express*, 2021, **29**, 6434–6444.
- 12 R. Xu, X. Liu and Y.-S. Lin, *Results Phys.*, 2019, **13**, 102176.
- 13 R. Xu and Y.-S. Lin, *Opt. Lett.*, 2018, **43**, 4783–4786.
- 14 R. Xu, J. Luo, J. Sha, J. Zhong, Z. Xu, Y. Tong and Y.-S. Lin, *Appl. Phys. Lett.*, 2018, **113**, 101907.
- 15 Z. H. Jiang, S. Yun, F. Toor, D. H. Werner and T. S. Mayer, *ACS Nano*, 2011, **5**, 4641–4647.
- 16 A. Lochbaum, Y. Fedoryshyn, A. Dorodnyy, U. Koch, C. Hafner and J. Leuthold, *ACS Photonics*, 2017, **4**, 1371–1380.
- 17 D. Hasan and C. Lee, *Adv. Sci.*, 2018, **5**, 1700581.
- 18 A. Lochbaum, A. Dorodnyy, U. Koch, S. M. Koepfli, S. Volk, Y. Fedoryshyn, V. Wood and J. Leuthold, *Nano Lett.*, 2020, **20**, 4169–4176.
- 19 T. Beni, N. Yamasaku, T. Kurotsu, N. To, S. Okazaki, T. Arakawa, A. Balcytics, G. Seniutinas, S. Juodkakis and Y. Nishijima, *ACS Sens.*, 2019, **4**, 2389–2394.
- 20 P. Pitchappa, C. P. Ho, P. Kropelnicki, N. Singh, D. L. Kwong and C. Lee, *Appl. Phys. Lett.*, 2014, **104**, 201114.
- 21 R. Xu and Y.-S. Lin, *Nano Lett.*, 2021, **21**(7), 3205–3210.
- 22 F. Zhu and Y.-S. Lin, *Opt. Laser Technol.*, 2021, **134**, 106635.
- 23 T. Xu, R. Xu and Y.-S. Lin, *Results Phys.*, 2020, **19**, 103638.
- 24 R. Xu, X. Xu, B.-R. Yang, X. Gui, Z. Qin and Y.-S. Lin, *Photonics Res.*, 2021, **9**(7), 1409–1415.
- 25 H. Wang, Y. Yang and L. P. Wang, *Appl. Phys. Lett.*, 2014, **105**, 071907.
- 26 L. Yang, P. H. Zhou, T. X. Huang, G. S. Zhen, L. Zhang, L. Bi, X. L. Weng, J. L. Xie and L. J. Deng, *Opt. Mater. Express*, 2017, **7**, 2767–2776.
- 27 G. Dayal, X. Y. Chin, C. Soci and R. Singh, *Adv. Opt. Mater.*, 2017, **5**, 1600559.
- 28 K. V. Sreekanth, R. Medwal, Y. K. Srivastava, M. Manjappa, R. S. Rawat and R. Singh, *Nano Lett.*, 2021, **21**(23), 10070–10075.
- 29 K. V. Sreekanth, C. M. Das, R. Medwal, M. Mishra, Q. L. Ouyang, R. S. Rawat, K. T. Yong and R. Singh, *Adv. Mater.*, 2021, **33**, 2006926.
- 30 K. V. Sreekanth, M. ElKabbash, R. Medwal, J. H. Zhang, T. Letsou, G. Strangi, M. Hinczewski, R. S. Rawat, C. L. Guo and R. Singh, *ACS Photonics*, 2019, **6**, 1610–1617.
- 31 K. V. Sreekanth, S. Han and R. Singh, *Adv. Mater.*, 2018, **30**(21), 1706696.
- 32 K. V. Sreekanth, S. Sreejith, S. Han, A. Mishra, X. X. Chen, H. D. Sun, C. T. Lim and R. Singh, *Nat. Commun.*, 2018, **9**, 369.
- 33 T. L. Wang, Y. P. Zhang, H. Y. Zhang and M. Y. Cao, *Opt. Mater. Express*, 2020, **10**, 369–386.
- 34 X. M. Tian and Z. Y. Li, *Plasmonics*, 2018, **13**, 1393–1402.
- 35 R. Z. Sun, P. H. Zhou, W. S. Ai, Y. N. Liu, Y. Li, R. M. Jiang, W. X. Li, X. L. Weng, L. Bi and L. J. Deng, *Opt. Express*, 2019, **27**, 11537–11546.
- 36 C. Y. Peng, K. Ou, G. H. Li, X. Y. Li, W. J. Wang, Z. Y. Zhao, X. S. Chen and W. Lu, *Opt. Express*, 2020, **28**, 11721–11729.
- 37 D. Li, H. L. Huang, H. Xia, J. P. Zeng, H. J. Li and D. Xie, *Results Phys.*, 2018, **11**, 659–664.
- 38 W. V. Hans, A. S. Barker and C. N. Berglund, *Phys. Rev.*, 1968, **40**(4), 737.
- 39 D. R. Smith, D. C. Vier, T. Koschny and C. M. Soukoulis, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2005, **71**, 036617.
- 40 H. Kim, K. Cheung, R. C. Y. Auyeung, D. E. Wilson, K. M. Charipar, A. Pique and N. A. Charipar, *Sci. Rep.*, 2019, **9**, 11329.