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# Nanomaterials for photocatalytic hydrogen production: from theoretical perspectives

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To overcome the increasing demand of energy worldwide and global warming due to CO2 emissions from the use of traditional fuel sources, renewable and clean energy sources are in high demand. Solar energy is one of the important renewable energy sources since it can be converted into hydrogen fuel via water splitting. To produce highly efficient and low-cost H2 from the reduction of water and CO<sub>2</sub>, photocatalytic materials have been developed. Presently, the efficiency of H<sub>2</sub> production using transition metal-oxide and non-metal oxide photocatalysts via water splitting is quite low. The main issues reported are low light absorption and poor charge separation. The reasons for these two issues are the large band gaps and band edge positions of the present photocatalysts used for H<sub>2</sub> production. To produce H<sub>2</sub> to a larger extent from the decomposition of water, the present photocatalysts have been modified by co-catalysts or dopants using different techniques, including the reduction of the band gap and adjustment of the morphology, band edge positions, crystallinity, surface structure, etc., such that these photocatalysts can absorb sufficient light in the visible-light region. This type of modified nanostructured photocatalysts (both oxide and non-oxide) can enhance the efficiency of H<sub>2</sub> production via absorbing sufficient light in the visible-light region of the solar spectrum and improving the charge separation by suppressing charge recombination. In this regard, we reviewed both UV- and visible-light active nanostructured photocatalysts and modified photocatalysts reported in theoretical studies.

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#### 1. Introduction

The use of traditional energy sources, such as fossil and mineral fuels and nuclear and hydroelectric sources, causes global warming due to the emission of  $CO_2$ , methane, CO, etc. that can cause serious disasters in various places worldwide. Moreover, there is a continuous depletion of traditional energy sources. To solve these issues, alternative energy sources, such as renewable energy sources, have been developed by researchers worldwide to meet the world's energy needs as well as to decrease pollution.

Various types of renewable energy sources are under development or have been developed. Among these, hydrogen can be a promising potential candidate and its production from renewable energy sources is CO<sub>2</sub>-free. In this regard, hydrogen can be considered as a clean energy carrier that can address the current energy and environmental issues.<sup>1,2</sup>

Hydrogen can be produced from primary energy sources including coal, natural gas, waste, biomass, solar, wind, hydropower, nuclear power, and geothermal power. For example, commercial hydrogen can be produced *via* steam reforming of methane<sup>3,4</sup> and also from coal using gasification

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technology; however, both these methods result in  $CO_2$  emissions. Hydrogen can be stored in its gaseous, liquid, or metal hydride form and can be distributed over large distances via pipelines or tankers. Numerous studies have been carried out to discuss the key role of hydrogen in sustainable development. <sup>5-10</sup>

The abundant supply of water and sunlight offers us an affordable alternative source to produce hydrogen. Therefore, photocatalytic water splitting is one of the potential techniques for clean solar hydrogen production and has been utilized in small- to large-scale hydrogen generators.11 In 1972, Honda and Fujishima<sup>12</sup> first investigated water splitting using a single TiO<sub>2</sub> crystal as a photoanode and Pt as a cathode. Subsequently, numerous studies on water splitting have been conducted13-19 with the exploration of more than 100 different new catalysts including multi-component oxides, sulfides, nitrides, and carbides. All of them are concentrated on the various factors affecting the conversion efficiency such as how much sunlight is absorbed, the exciton generation, the separation/recombination of holes and electrons received by dissociation of excitons, and the charge collection by the respective electrodes to produce hydrogen and oxygen. Some other practical factors are also important for photocatalytic hydrogen production such as stability and resistance under visible light, reduced cost, and non-toxicity of utilized materials.<sup>20-23</sup> In this regard, we speculated that photocatalytic hydrogen could be a commercial fuel **RSC Advances** Review

in future; however, significant research efforts should be devoted in this direction, addressing the abovementioned key issues for its industrial production.

Nanomaterials such as CdS, SiC, CuInSe2, and TiO2 can be utilized to produce cheap and clean photocatalytic hydrogen<sup>24-27</sup> due to the fact that these nanomaterials show better photocatalytic properties as compared to their bulk counterparts. Currently, many other nanomaterials such as Nb<sub>2</sub>O<sub>5</sub>, <sup>28</sup> Ta<sub>2</sub>O<sub>5</sub>, <sup>29</sup> α-Fe<sub>2</sub>O<sub>3</sub>, $^{30,31}$  ZnO, $^{32,33}$  TaON, $^{34}$  BiVO<sub>4</sub>, $^{35,36}$  and WO<sub>3</sub> have been explored.37 In most of the photocatalysts, band gap limitation is a key issue that results in low H2 production.38 To solve this issue, some photocatalysts have been modified via noble metal doping, ion doping, sensitization, and metal-ion implantation. In noble metal doping, Pt has been found to be the best noble metal; however, it is too expensive. Thus, other efficient and cheaper metals such as Ni, Cu, Ag, Ru, Pd, and Ir have been explored.39-47 Ion doping involves transition metal and rareearth metal ions, anions of nitrogen and sulphur, etc. Sensitization means dye-sensitization and coupling of semi-Among the abovementioned conductors. modification techniques, metal-ion implantation and dye-sensitization have been found to be the most effective photoanode surface modification techniques. Moreover, researchers are interested in exploring co-catalysts with photocatalyst nanomaterials that can be used for photocatalytic hydrogen production.

The theoretical and computational description of many body systems is still one of the biggest challenges in solid state science although significant progress has been made in this regard. The majority of theoretical and computational studies on photocatalytic nanomaterials have used density functional theory (DFT) because DFT predicts the ground-state properties in terms of electronic density. The band gap of the materials is often significantly underestimated by DFT,48 computed using Kohn-Sham method by introducing an exchange correlation (XC) functional such as local density approximation (LDA)49 or the generalized gradient approximation (GGA).50 The band gap of the materials can be improved by applying several approaches. For example, a hybrid functional can improve the accuracy by combining the approximate DFT functionals (LDA or GGA) with the exact Hartree-Fock (HF) exchange energy.<sup>51</sup> Alternatively, the DFT + U method with the Hubbard band U term can provide more accurate results.52 The accuracy of these two approaches depends on the selection of several empirical parameters. Moreover, the parameter-free GW method yields an accurate description of the electronic structure of the materials, but at a huge computational expense.53 Therefore, the use of GW method is not more reliable for complex structures.

Herein, the study was concentrated on modeling a nanostructured photocatalyst and modified nanophotocatalysts at the atomic level to provide information on the key atomic level structures, processes, and parameters that control the behavior of photocatalytic materials during their applications in photocatalytic H<sub>2</sub> production. In particular, substantial contributions have been made by modeling methods in the study of the electronic states, defects, and surface properties of nanophotocatalysts for H2 production.

# Theoretical achievements for the development of nanophotocatalysts and modified photocatalysts for hydrogen production via water splitting

Many nanophotocatalysts have been developed to split water into H2 and O2 under UV and visible-light illumination. The industrial production of clean and recyclable H2 via the direct splitting of water using a particulate photocatalyst could be the best method.54-57 However, there is a lack of suitable materials with appropriate band gap positions for overall water splitting, and stability is required for practical applications. The reason is that most metal oxides have optical band gaps that lie outside the visible-light range ( $E_g > 3$  eV); thus, even if they are catalytically active for a given reaction, they cannot make efficient use of the solar spectrum. For example, TiO2 has a wide band gap ( $E_g = 3.2$  eV), which is limited to a 1% solar-to-hydrogen (STH) conversion efficiency for the water splitting process, whereas for a smaller band gap material such as  $Fe_2O_3$  ( $E_g=2.2$ eV), the theoretical efficiency increased to 15%.58 The STH of 10% is required for the solar cell-powered electrolysis of water. Therefore, the development of highly active photocatalysts should be explored for large-scale photocatalytic H2 production via water splitting.59-62 There are two types of nanostructured photocatalyst materials for H2 production via water splitting.

#### (i) Metal-oxide photocatalysts and modified metal-oxide photocatalysts (with hybrid systems)

Although there have been many experimental investigations reported on metal oxide nanophotocatalysts for H<sub>2</sub> production via water splitting under UV- and visible-light irradiation, very few theoretical investigations based on the DFT methods have been reported. Among these, most theoretical investigations have been reported on nanostructured TiO2 and modified TiO2 photocatalysts. For example, Kaur et al.63 investigated amorphous TiO2 as a photocatalyst for H2 production using the DFT methods. The authors claimed via the analysis of the electronic properties that amorphous TiO2 may act as a cheaper, more abundant, but somewhat less efficient photocatalyst as compared to crystalline TiO2. To improve the photocatalytic activity of the TiO2 nanophotocatalyst, the doping of TiO2 was carried out using different techniques via narrowing the band gap of TiO<sub>2</sub> to absorb more visible-light in the solar spectrum as well the dependence of charge separation and recombination on the distance between the catalytic core and semiconductor surface. For example, Reynal et al.64 reported that the photoinduced reduction of H2 on a Co electrocatalyst immobilized on TiO<sub>2</sub> was 10<sup>4</sup> times faster than the reverse charge recombination. The authors claimed that both processes show an exponential dependence on the distance between the semiconductor and the catalytic core, as shown in Fig. 1(a)-(d).64 As shown in Fig. 1, the authors computed the charge separation and recombination processes when a semiconductor (TiO2) was functionalized with three related cobalt electrocatalysts, whose

Fig. 1 (a) The electron transfer processes in  ${\rm TiO_2}$  functionalised with a molecular catalyst for  ${\rm H_2}$  production after UV-light excitation. The solid black and dashed red arrows indicate the charge separation and recombination, respectively. The molecular structures of the catalysts for  ${\rm H_2}$  reduction are shown in (b) for Co1, (c) for Co2, and (d) for Co3 (the charges have been omitted for clarity). The blue arrows indicate the distance between the anchoring groups and the catalyst metal centre (r, A), as determined by the energy minimised DFT calculations. Reprinted with permission of ref. 64. Copyright 2014 Royal Society of Chemistry.

molecular structure varied the physical separation between the catalytic core and the semiconductor surface. In this hybrid system, the semiconductor acted as a light harvester and H2 evolution was driven by the anchored molecular catalyst. Some theoretical investigations have shown that BaTiO<sub>3</sub> (ref. 65 and 66) and PbTiO<sub>3</sub> (ref. 67-69) nanowires exhibit ferroelectricity and Bi<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>, o SrTiO<sub>3</sub>, and CaTiO<sub>3</sub> (ref. 72) nanowires exhibit photocatalytic activity towards water splitting. Bandura et al.73 demonstrated the photocatalytic activity of SrTiO<sub>3</sub> nanowires and concluded that via comparison of the band edge positions and Fermi levels relative to the levels of the reduction H<sup>+</sup>/H<sub>2</sub> and oxidation O2/H2O potentials as well as analysis of the band gap widths, the probable SrTiO3 nanowire configurations and SrTiO<sub>3</sub> nanotube<sup>74</sup> configurations can be suggested for application in the photocatalytic splitting of water molecules under solar irradiation in the visible-light range.

In the case of TiO<sub>2</sub>, the phase of the photocatalyst is a key factor for its photocatalytic activity. For example, although the rutile phase exhibits a narrower band gap (approx. 3.0 eV) than the anatase phase (approx. 3.2 eV),75-77 anatase is generally considered to exhibit superior photocatalytic activity due to its larger surface area and thus higher activity.<sup>78</sup> The third metastable phase is brookite that has also been reported to exhibit photocatalytic activity, 79,80 and mixtures of anatase, rutile, and brookite have been reported to exhibit significant photocatalytic activity.81,82 However, this phase is of lesser interest due to the complexity of its synthesis. The doping of TiO2 can provide several outcomes: formation of new valence states in TiO2, 83-86 creation of charge carrier trapping sites,87-90 band gap reduction,91-93 control of phase transformation behavior,94-97 and surface enhancement. 96,98 The effects of dopants on the anatase to rutile phase transformation have been comprehensively reviewed elsewhere.99 The phase transformation mainly occurs due to the atomic rearrangement involved in the transformation, which is often a result of an increase in the density of the anion vacancies. This may occur when cationic dopants of low valence substitute Ti in the anatase lattice.79,99,100 Hanaor et al. 100 performed DFT calculations for the relative stability of anatase and rutile polymorphs of TiO2. The rutile phase exhibited a stability of 3 meV as compared to the anatase phase in pristine TiO2. The authors also demonstrated the doping of TiO2 with Si, Al, Fe, and F atoms and found that the anatase phase stabilized the rutile phase in the dopant order F > Si > Fe > Al obtained *via* comparison of  $\Delta F$  (a-TiO<sub>2</sub>:F) with  $\Delta F$  (r-TiO<sub>2</sub>:F), which provided their formation energies as 84.78, 32.43, 18.46, and 17.68 meV, respectively. As the authors considered only flourine anion dopant in their study, the inhibition of rutile formation via F doping was supported by previously reported experimental results. 101,102 A schematic of the interstitial and substitutional dopant positions in anatase and rutile TiO2 is shown in Fig. 2(a) and (b).100

In general, efforts to modify TiO<sub>2</sub> to enhance the visible-light absorption have been focused on the substitution of metal cations and/or non-metal anions. 103-111 In some cases, the choice of DFT methodology can have an influence on the reduction of the band gap to the visible-light region of the solar spectrum. 109 Co-doping with compensating cation-anion pairs from DFT simulations has been reported, 112,113 along with a recent experimental study of Mo-C co-doped TiO2.114 Key parameters such as stability, solubility, and reproducibility are also important along with the band gap reduction as well as the method of doping and/or the DFT methodology to enhance the photocatalytic activity of TiO2 due to minimum charge recombination. Moreover, nanostructure engineering of TiO2 can reduce the charge recombination. 115-119 In this regard, heterostructure techniques of TiO2 with other oxides can enhance the photocatalytic activity to tune the structure to increase the absorption of visible-light. For example, heterostructures such as Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>-TiO<sub>2</sub>, <sup>124</sup> MgO-TiO<sub>2</sub>, <sup>134</sup> and Ga<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> (ref. 134) can reduce the band gap, leading to visible-light photoactivity, efficient charge separation, and improved photocatalytic activity as compared to pure TiO2.

In the case of the  $\rm Bi_4Ti_3O_{12}$ – $\rm TiO_2$  heterostructure, Bi 6s mainly contributes to the valence band, whereas the Bi 6p state contributes to the conduction band, which result in the reduction of the band gap of  $\rm Bi_4Ti_3O_{12}$  (2.5 eV) as compared to that of  $\rm TiO_2$  (3.2 eV). This leads to large absorption of visible-

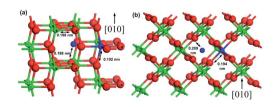


Fig. 2 A schematic of the interstitial and substitutional dopant positions in (a) anatase and (b) rutile  $TiO_2$  with the arrows showing the cation-dopant to oxygen distances. Reprinted with permission of ref. 100. Copyright 2012 Springer-Verlag Berlin Heidelberg.

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light. A schematic showing the energy band structure and electron-hole pair separation in the Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>-TiO<sub>2</sub> heterostructure is shown in Fig. 3.124 Similarly, in the case of MgO-TiO<sub>2</sub> and Ga<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> heterostuctures, the author claimed the presence of two states in the band gap, as shown in Fig. 4. 134 The first state comes from an occupied Ti<sup>3+</sup> state at 0.9 eV and 1.0 eV above the valence band edge in the MgO-TiO2 and Ga2O3-TiO2 heterostuctures, respectively. The second state comes from an unoccupied O 2p state (oxygen hole polaron), which lies 1.8 eV and 1.5 eV above the valence band of the MgO-TiO2 and Ga2O3-TiO<sub>2</sub> heterostuctures, respectively. These results show that the excitonic electrons and holes are localized on the TiO2 and the metal oxide nanostructures, which lead to a reduction in charge recombination and an enhancement in the photocatalytic

In addition, TiO2 nanoparticles can attach materials such as zeolites, 135 silica, 136 activated carbon 137-142 or carbon nanotubes. 143-149 Recently, graphene has attracted significant interest150-154 due to its large surface area and potentially higher photocatalytic efficiency.<sup>155</sup> However, very few reports are available for theoretical investigations on the interaction of nanoparticles and graphene sheets, and the mechanism of charge transfer from graphene to the nanoparticles. The decorated surface of graphene with both TiO2 nanofibers155 and Fedoped TiO2 nanofibers156 in supercritical carbon dioxide exhibits high photocatalytic activity. Das et al. 157 deposited different semiconductor nanoparticles, such as TiO2 and ZnO, and some magnetic nanoparticles, such as Fe<sub>3</sub>O<sub>4</sub> and Ni, on graphene. They calculated their electronic structure using DFT methods and found that charge transfer occurred between graphene and the deposited nanoparticles.

Among all the metal oxides, WO<sub>3</sub> has attracted significant interest due to its photosensitivity, 158-160 good electron transport properties,161 and stability against photocorrosion.162,163 Moreover, its smaller band gap (approx. 2.8 eV) than other oxides such as TiO<sub>2</sub> makes it suitable for the absorption of visible-light. However, the gap of WO<sub>3</sub> is still too large for sufficient visiblelight absorption. Experimental results have shown that the conduction band minimum (CBM) of bulk WO3 is about 0.4 eV below the hydrogen redox potential. 164,165 For the WO3 surface, the CBM lies 0.31 eV above the hydrogen redox potential, which is still low for hydrogen production.166

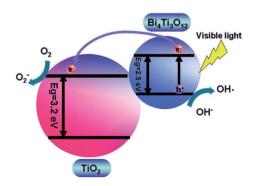


Fig. 3 A schematic showing the energy band structure and electronhole pair separation in the Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>/TiO<sub>2</sub> heterostructure. Reprinted with permission of ref. 124. Copyright 2011 Royal Society of Chemistry.

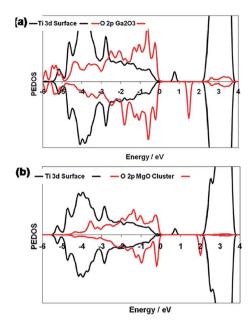


Fig. 4 Ti 3d and O 2p PEDOS of the (Ga<sub>2</sub>O<sub>3</sub>)<sub>2</sub> and (MgO)<sub>4</sub> clusters for the relaxed triplet excited states of (a) Ga<sub>2</sub>O<sub>3</sub>-modified TiO<sub>2</sub> and (b) MgO-modified TiO<sub>2</sub>. The zero of energy is the top of the valence band. Reprinted with permission of ref. 134. Copyright 2012 American Chemical Society.

Doping and co-doping are good ways to tailor the electronic structure and photocatalytic properties of semiconducting oxides. For example, Mo-doped WO3 nanowires showed an enhancement in visible-light photoactivity, as the band gap of the Mo<sub>x</sub>W<sub>1-x</sub>O<sub>3</sub> solid solutions was narrowed by 0.48 eV upon increasing the Mo content from 0 to 0.75. 167,168 The effect of several transition metals on the photocatalytic activity under UV irradiation was also studied, suggesting maximum H2 production via Ni doping with WO<sub>3</sub>.<sup>169</sup> Doping with other metals (Ti, Zn, Dy, Te, Ta, V, Cu, Ag, and Ce) has also been reported in the literature, 170-178 with some claims of improved photocatalytic activity. For example, Wang et al. 179 studied Mo, Cr, Ti, Zr, and Hf-doped WO3 and found that in Hf-doped WO3, the oxygen vacancies have a negative formation energy, which leads to a shift in both the VB and CB to higher energies and a reduction of the band gap, with potential benefits for photocatalytic H<sub>2</sub> production, as shown in Fig. 5.179

In previous studies, it has been reported that perovskite-type alkali tantalates, ATiO<sub>3</sub> (A = Li, Na, and K) show significant photocatalytic activity under UV irradiation. However, these are less active under visible-light irradiation. 180,181 Transition metal ions were found to be attractive dopants for the effective modification of photocatalysts, which can play multiple roles ranging from reducing the band gap to efficient electron transfer due to their multiple oxidation states. 182,183 Electron transitions from the valence band to the dopant level or from the dopant level to the conduction band can effectively red shift the band edge absorption threshold.

For example, Liu et al. 184 demonstrated M-doped NaTaO3 (M = V, Cr, Mn, Fe, Ni, Cu, and Zn) nanoparticles using both experimental and DFT techniques. The authors found that the Review RSC Advances

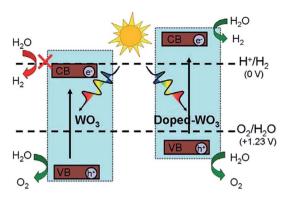


Fig. 5 A schematic of the superior photocatalytic activity of metal-doped  $WO_3$  for  $H_2$  production as compared to that of  $WO_3$ . Reprinted with permission of ref. 179. Copyright 2012 American Chemical Society.

substitution of Ta<sup>3+</sup> by metal ions results in the formation of an intermediate band, which is due to the metal 3d state. The supercell of M-doped NaTaO<sub>3</sub>, total and projected DOS of pure and V-doped NaTaO<sub>3</sub>, and intermediate band formation by the

d-orbitals of the doping metal ions are shown in Fig. 6(a), (b) and (c), respectively.<sup>184</sup>

Moreover, a ZnO/graphene hybrid nanostructure was reported to have excellent potential for application in electronic devices. 
<sup>185–190</sup> Most studies on the ZnO/graphene hybrid structures have been focused on their structural morphology and electronic properties. 
<sup>191,192</sup> Recently, Xiang *et al.* 
<sup>193</sup> first proposed a three-component composite, TiO<sub>2</sub>/graphene/MoS<sub>2</sub>, containing 0.25 wt% graphene. The results showed that graphene acts as an electron reservoir and MoS<sub>2</sub> can act as a source of active adsorption sites to achieve a highly efficient synergetic H<sub>2</sub> evolution at the rate of 165.3 µmol h<sup>-1</sup>, which is 17 times that of TiO<sub>2</sub>/MoS<sub>2</sub>. 
<sup>194</sup> Wu *et al.* 
<sup>195</sup> studied ZnO-dotted porous ZnS cluster microspheres for highly efficient, Pt-free photocatalytic H<sub>2</sub> evolution.

# (ii) Non-metal oxide photocatalysts and modified non-metal oxide photocatalysts (with hybrid systems)

The surface modification of oxide photocatalysts, particularly TiO<sub>2</sub>, *via* doping with N, C or S, or metal nanoparticles cannot provide significant photocatalytic activity in the modified

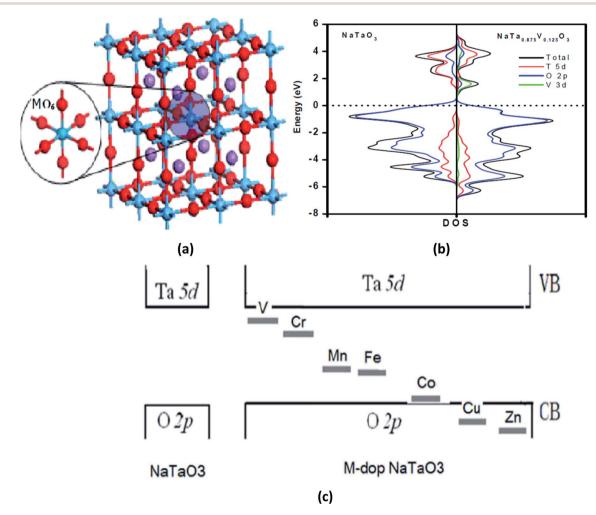


Fig. 6 (a) A supercell of M-doped NaTaO $_3$  2 × 2 × 2 (40-atoms), (b) the total density of states and projected density of states of the pure and V-doped NaTaO $_3$ , and (c) intermediate band formation by the d-orbital of the doped metal ions. Reprinted with permission of ref. 184. Copyright 2012 IACSIT Press, Singapore.

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materials in the visible-light region of the solar spectrum. 196,197 Therefore, researchers have been interested in exploring alternative materials for nanophotocatalysts and modified photocatalysts for photocatalysis and solar energy conversion. As a result, the large-scale production of stable visible-light active photocatalysts remains a challenge for industrial applications.198 For example, polymeric carbon nitride has been found to be an efficient photocatalyst that produces H<sub>2</sub> from water under visible-light irradiation; however, in this case, a sacrificial donor is required.199 Similarly, a class of metal-free photocatalysts, including elemental boron,200 sulfur,201 and phosphor,202 as well as the binary carbon nitride203 and boron carbide, has emerged.204 Conjugated polymer nanostructures have emerged as alternate materials for applications in solar energy conversion. However, there is a lack of photocatalytic studies of these conjugated polymers. Semiconductor nanostructures modified with conducting polymers have been studied for photocatalytic applications. Two-dimensional materials such as graphene<sup>205</sup> and layered hexagonal (h-BN) are promising photocatalysts. 206-208 However, graphene (monolayer) exhibits a zero band gap, whereas h-BN exhibits a wide band gap (~5.5 eV). These materials (say ternary B-C-N compounds) can constitute the desired medium band gap semiconductors via adjustment of the band gap and absolute energy levels via chemical variations. 209,210 Similarly, graphene oxide based on a carbon support may improve the charge separation and spontaneous redox processes.211,212 In recent years, graphene-based heterogenous photocatalytic

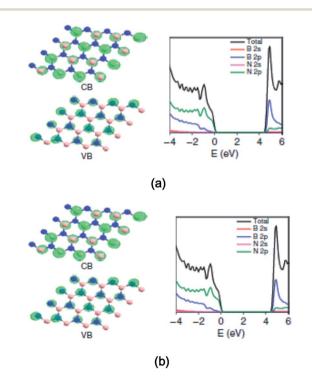


Fig. 7 The plane-wave DFT calculations of the electronic structure of h-BN and h-BCN. The optimized structure of B<sub>16</sub>N<sub>16</sub> with the corresponding valence band (VB)/conduction band (CB) and the corresponding total and ion-decomposed electronic density of states (a and b). Reprinted with permission of ref. 215. Copyright 2015 Macmillan Publishers Limited

nanomaterials have attracted significant attention due to their unique sp<sup>2</sup> hybrid carbon networks, exhibiting ultra-fast electron mobility at room temperature, conductivity, large theoretical surface area, high work function, etc.213-215 Huang et al.215 demonstrated a ternary structure of B-C-N nanosheets that has the functionality to catalyze H2 and O2 evolution from water as well as CO2 reduction under visible-light illumination. The plane-wave DFT calculations of the electronic structure of h-BN are shown in Fig. 5(a) and (b).215 As shown in Fig. 7(a) and (b), for a B<sub>11</sub>C<sub>12</sub>N<sub>9</sub> compound, the gap is significantly reduced from 4.56 eV to 2.00 eV. The partial DOS, as shown in Fig. 7(b), signifies that the VB and CB edges of B<sub>11</sub>C<sub>12</sub>N<sub>9</sub> mainly comprise C 2p orbitals, which is different from that of pure h-BN (Fig. 7(a)).

Metal chalcogenides have been found to be promising photocatalyst materials for photocatalytic H2 production due to their appropriate band gap width and band edge position.<sup>216</sup> In particular, CdS has been found to be an efficient semiconductor H<sub>2</sub>-production photocatalyst.<sup>217</sup> However, some key issues such as the quick recombination of photogenerated charge carriers and photocorrosion under visible-light irradiation still exist and prohibit the wide application of CdS to a large extent. To solve these issues, forming Zn<sub>1-x</sub>Cd<sub>x</sub>S solid solutions is a viable method because ZnS possesses the same coordination mode with CdS, 218-220 and the band gap width and band edge position of the Zn<sub>1-x</sub>Cd<sub>x</sub>S solid solutions can be tuned by changing the molar ratio of ZnS and CdS. 221,222 For example, Li et al. 223 demonstrated the visible-light photocatalytic H<sub>2</sub> production activity of Zn<sub>1-x</sub>Cd<sub>x</sub>S solid solutions with different molar ratios of ZnS and CdS using both experimental and DFT methods. The authors found that for Zn/Cd with a molar ratio equal to 1:1, the Zn<sub>0.5</sub>Cd<sub>0.5</sub>S solid solution exhibits the highest H<sub>2</sub>-production rate of 7.42 mmol h<sup>-1</sup> g<sup>-1</sup>, exceeding that of the pure CdS and ZnS samples by more than 24 and 54 times, respectively. The supercell models for bulk ZnS and CdS, the geometry of Zn<sub>0.5</sub>Cd<sub>0.5</sub>S, the band structures of the Zn<sub>1-x</sub>Cd<sub>x</sub>S systems for various Zn/Cd molar ratios and the conduction and valence band edge potentials of the samples (x = 0, 0.1, 0.3, 0.5, 0.7, 0.9,and 1.0) are shown in Fig. 8(a), (b) and (c), respectively.223 Fig. 8(d) shows that the band gap of the  $Zn_{1-x}Cd_xS$  samples gradually becomes narrower from 1.921 to 1.141 eV upon increasing the Cd content, which is well consistent with the experimental results.

Many studies have been conducted on the development of semiconducting material/graphene hybrid structures using either gas phase<sup>223-226</sup> or liquid phase techniques.<sup>185,186,191,227</sup> Hou et al.228 have developed a CdS QDs/graphene/ZnIn2S4 system that exhibits highly efficient H<sub>2</sub> production due to the high hydrothermal stability and efficient electron transfer. Graphene can also play a role as a co-catalyst and constitutes a synergistic effect with the other co-catalyst to improve the photocatalytic efficiency.<sup>229-231</sup> Similarly, Zhu et al.<sup>230</sup> demonstrated ZnS loaded with 0.25 wt% graphene and 2 atom% MoS<sub>2</sub> that yielded a high  $H_2$  production rate of 2258  $\mu$ mol  $h^{-1}$ , which is 2 times of that observed with ZnS only.

Consequently, there is a huge scarcity of theoretical investigations based on the DFT methods on nanophotocatalysts (both

(a) ZnS (b) CdS (c)  $Zn_{0.5}Cd_{0.5}S$  (b) CdS (c)  $Zn_{0.5}Cd_{0.5}S$  (b) CdS (c)  $Zn_{0.5}Cd_{0.5}S$  (c)  $Zn_{0.5}Cd_{0.5}S$ 

Fig. 8 The supercell models for bulk (a) ZnS (64 atoms) and (b) CdS (64 atoms). The red, yellow, and blue spheres represent the Zn, S, and Cd atoms, respectively. (c) The geometry model for  $Zn_{0.5}Cd_{0.5}S$  and (d) conduction and valence band edge potentials of the  $Zn_{1-x}Cd_xS$  samples (x = 0, 0.1, 0.3, 0.5, 0.7, 0.9, and 1.0). Reprinted with permission of ref. 223. Copyright 2013 American Chemical Society.

metal-oxide and non-metal oxide) and modified photocatalysts for photocatalytic H<sub>2</sub> production via water splitting. In the case of metal-oxide photocatalysts, few studies were reported on pure TiO<sub>2</sub> and modified TiO<sub>2</sub>, which were focused on the electronic structures of modified TiO2, signifying the gap reduction and efficient charge separation due to doping of TiO2 to enhance the photocatalytic H<sub>2</sub> production. Moreover, very few studies have been reported for graphene and metal chalcogenides in the form of B-C-N ternary compounds and multicomponent heterostructures (metal oxide/graphene/metal chalcogenides) for efficient photocatalytic H2 production due to the synergistic effect of graphene and MoS<sub>2</sub>. There is a big opportunity for theoretical researchers to conduct DFT calculations on metal oxide and nonmetal oxide photocatalysts and their modified forms for highly efficient photocatalytic H2 production. Therefore, more DFT investigations should be devoted to either experimentally explore photocatalysts and modified photocatalysts or on the exploration of new photocatalytic materials for highly efficient H<sub>2</sub> production to a large extent.

## 3. Summary and perspectives

Herein, we reviewed nanostructured photocatalysts (both transition metal oxides and non-metal oxides) and modified photocatalysts for highly efficient  $H_2$  production via water splitting under both UV and visible-light irradiation investigated via DFT techniques. However, the current lack of industrial applications of current semiconductor-based photocatalytic  $H_2$  production is mainly due to two reasons: the low photocatalytic efficiency and lack of extensive studies for successful industrial applications of photocatalytic  $H_2$  production experimentally as well as the lack of DFT studies to investigate the factors responsible for the enhancement of the efficiency of  $H_2$  production. The current  $H_2$  production yield is quite low, which is far from the targeted quantitative efficiency of 30% at 600 nm for practical applications. The critical light conversation efficiency for photocatalytic  $H_2$  production via water splitting is about 15%.  $^{233}$  In

this regard, the present research does not explore the highly efficient and cost-effective photocatalysts and modified photocatalysts for H2 production via water splitting. The reason is that the present metal oxide and non-metal oxide photocatalysts have wide band gaps, which do not absorb sufficient sunlight in the visible-light region. To reduce the band gaps towards visible-light absorption and photogenerated charge separation in photocatalysis, the photocatalysts have been modified using different techniques in different forms. However, the modified photocatalysts do not enhance the efficiency of H2 production from the decomposition of water under visible-light irradiation. Moreover, the fast charge recombination and fast back reactions are also drawbacks for the solar water splitting system.234-236 DFT investigations have reported the control of band gap and band edge positions via variations of the morphology, composition, crystallinity, and surface structure of photocatalysts for efficient H2 or O2 evolution from water splitting.

Consequently, more theoretical studies should be devoted towards the exploration of highly efficient, stable, eco-friendly and cost-effective photocatalysts, and modified photocatalysts via the continual addition of electron donors to overcome the key issues of current solar water splitting systems in the production of  $H_2$  to a larger extent.

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