



Cite this: *J. Mater. Chem. C*, 2023, 11, 9144

Multifunctional pseudohalide-based ionic liquid doping promotes efficient and stable perovskite solar cells†

Xing Guo,^a Zhenhua Lin,^b Wenyong Cao,^b Yumeng Xu,^b Qingrui Wang,^b Boyao Zhang,^b Yue Hao^b and Jingjing Chang^b

High-quality perovskite films are crucial to realizing high-efficiency stable perovskite solar cells. Doping is a direct and effective strategy to passivate defects and improve the crystal quality of perovskite films. Herein, we introduce pseudohalide-based ionic liquid TEAPF₆ into perovskite films as an excellent additive. TEAPF₆ doping has multiple effects: the PF₆⁻ anion could fill the halide vacancies, which helps to suppress ion migration through vacancies. The highly electronegative PF₆⁻ could bind with uncoordinated Pb²⁺ and organic cations to passivate defects. TEAPF₆ doping improved the crystal quality and suppressed the charge recombination in perovskite films. More importantly, the TEA⁺ with long alkyl groups and PF₆⁻ could enhance the hydrophobicity of the perovskite films. As a result, the device with TEAPF₆ achieved a PCE of 22.13%. Moreover, TEAPF₆ doping enhanced the stability of PSCs. This work demonstrates the excellent effect of pseudohalide-based ionic liquid doping on improving the crystal quality of perovskite films.

Received 23rd March 2023,
Accepted 30th May 2023

DOI: 10.1039/d3tc01028g

rsc.li/materials-c

1. Introduction

Over the past few years, metal halide perovskites have been extensively studied due to their tunable bandgap, strong light absorption, long carrier diffusion length, and low-temperature solution processing.^{1–5} Since 2009, the power conversion efficiency (PCE) of perovskite solar cells (PSCs) has rocketed from 3.8% to 25.7%.^{6–12} This efficiency has already met the requirements of commercial applications, but it is still far from

^a Advanced Interdisciplinary Research Center for Flexible Electronics, Academy of Advanced Interdisciplinary Research, Xidian University, 710071, Xi'an, China.
E-mail: zhlin@xidian.edu.cn, jjingchang@xidian.edu.cn

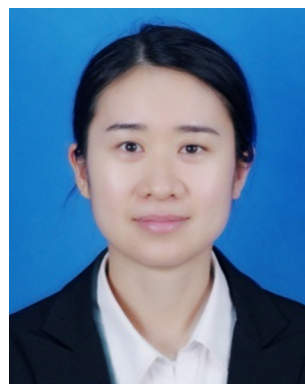
^b State Key Discipline Laboratory of Wide Band Gap Semiconductor Technology, School of Microelectronics, Xidian University, 710071, Xi'an, China

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3tc01028g>



Xing Guo

Dr Xing Guo obtained his PhD degree in Microelectronics and Solid State Electronics from Xidian University in 2021. In 2021, he joined Academy of Advanced Interdisciplinary Research, Xidian University as a lecturer. His current research focuses on organic-inorganic hybrid perovskite solar cells, and energy conversion and storage device integration.



Zhenhua Lin

Prof. Zhenhua Lin received her PhD degree from the National University of Singapore in 2013. After graduation, she was a Research Fellow in Electrical and Computer Engineering, National University of Singapore from 2013 to 2014. She joined Xidian University, where she is currently a professor. Her research interests include perovskite and metal oxide materials, electronic devices (transistors and RRAM) and optoelectronic devices (solar cells, photodetectors, and X-ray detectors).

reaching the theoretical limit of efficiency.^{13,14} It also continues to be challenging for PSCs to achieve the necessary stability for commercialization. High-quality perovskite films are the key to improving the efficiency and stability of PSCs. It is well known that perovskite films prepared by the solution process are polycrystalline, and contain a large number of defects. The defects as the recombination center will cause nonradiative recombination of the carrier. On the other hand, ion migration between defects under an electric field also causes perovskite decomposition. Therefore, defect passivation of perovskite films is an effective strategy to further improve the efficiency and stability of PSCs.^{15–19}

Ionic liquids (ILs) are a class of low-melting salts composed of anions and cations, and can achieve different properties by adjusting the combination of anions and cations. ILs have been widely used in perovskite defect passivation due to their excellent electrical conductivity and stability.^{20–26} Liu *et al.* doped CsPbI₃ perovskites with 1-ethyl-3-methylimidazolium hydrogen sulfate (EMIMHSO₄) ILs to manage the defects, energy alignment, and stability of perovskite films fabricated *via* blade-coating.²⁷ The PSCs doped with EMIMHSO₄ achieved a PCE of 20.01%, and the encapsulated PSCs retained 95% of their initial PCE after 1000 hours under ambient conditions. Akin *et al.* introduced 1-hexyl-3-methylimidazolium iodide (HMII) ILs as an additive into FAPbI₃ perovskites.²⁸ They found that HMII ILs could facilitate the grain coarsening of FAPbI₃ crystals and reduce the grain-boundary migration activation energy. The grain size of HMII-doped perovskites reached the micrometer scale. The PSCs with HMII achieved a PCE of 20.6% with a notable increased open-circuit voltage (V_{OC}) of about 80 mV. Moreover, the HMII-doped PSCs retained over 80% and about 90% of their initial PCE at 60 ± 10% relative humidity and 65 °C, respectively. Due to the high chemical activity, the I[−] in perovskite films is prone to form vacancies and other defects. The defect density, crystal quality, and stability of perovskite films can be effectively regulated by introducing other halide anions such as Br[−] and Cl[−].²⁹ Besides, some pseudohalide anions with similar properties to halide anions, such as BF₄[−], PF₆[−], SCN[−], NO₃[−], and CH₃COO[−], are also introduced into perovskites.^{20,30–32} Among them, BF₄[−] and PF₆[−] not only have a similar ionic radius to I[−] but also have stronger electronegativity than single halide anions, which could substitute the I[−] and suppress the defects related to halides. ILs with pseudohalide anions can combine multiple advantages to improve the properties of perovskite films. Snaith *et al.* incorporated 1-butyl-3-methylimidazolium tetrafluoroborate (BMIMBF₄) ILs into perovskites thereby increasing the efficiency and long-term stability of PSCs.³³ The most stable packaged PSCs degraded by only about 5% over 1800 hours of continuous simulated full-spectrum sunlight at 70 to 75 °C, and they also estimated that it will take about 5200 hours for the PSCs to drop to 80% of their peak performance. Furthermore, they doped perovskite films with BMPBF₄ ILs and found that BMPBF₄ ILs could retard impurity phase inversion and pinhole formation during aging.³⁴ The unencapsulated and encapsulated PSCs retained 80% and 95% of their peak PCE

for 1010 and 1200 hours under full-spectrum simulated sunlight in an ambient atmosphere.

In this work, we introduced a pseudohalide-based IL tetraethylammonium hexafluorophosphate (TEAPF₆) into Cs_{0.05}MA_{0.68}FA_{0.27}PbI_{3−x}Cl_x perovskite films as a multifunctional additive. TEAPF₆ contains a large structurally stable organic cation (TEA⁺) and a pseudohalide anion (PF₆[−]). The PF₆[−] could fill the halide vacancies and bond with Pb²⁺ and organic cations. The TEA⁺ with long alkyl groups and PF₆[−] could enhance the hydrophobicity of the perovskite film. The TEAPF₆ additive has an excellent effect on improving the crystal quality and effectively passivating the defects of perovskite films. As a result, the PSCs with TEAPF₆ achieved the highest PCE of 22.13% with a significantly increased V_{OC} from 1.10 V to 1.15 V. Moreover, TEAPF₆ ILs increased the stability of the PSCs.

2. Results and discussion

Fig. S1 (ESI[†]) shows the chemical structure of the TEAPF₆ IL, and the TEAPF₆ contains an organic cation (TEA⁺) and a pseudohalide anion (PF₆[−]). The TEA⁺ cation has four alkyl groups, which are larger than MA⁺, FA⁺, and Cs⁺ cations in perovskites. The pseudohalide anion PF₆[−] has similar properties and ionic radius to I[−] in the perovskite. The perovskite film acts as both a light absorption layer and a charge transport layer, and its surface morphology, crystalline quality, and other characteristics have a great impact on the efficiency and stability of PSCs. We first clarified the effect of TEAPF₆ doping on the morphology of perovskite films, and we measured the top-view and cross-sectional scanning electron microscopy (SEM) images of perovskite films without and with TEAPF₆. As shown in Fig. 1a–d and Fig. S2 (ESI[†]), the grain size of the perovskite films without and with TEAPF₆ is similar. The pristine perovskite film shows wider grain boundaries, and the cross-section of the pristine perovskite film is unordered and has some holes, which may lead to charge recombination and device leakage, and it is also easier for water and oxygen to enter the interior and cause the decomposition of the perovskite film. While the perovskite films with TEAPF₆ have denser grains and smaller grain boundaries, the cross-section exhibits more compact and vertically aligned crystal growth, which is beneficial to reducing the charge recombination at the grain boundaries and suppressing the decomposition of the perovskite film.

We further characterized the effect of the TEAPF₆ additive on the crystal quality of perovskite films by X-ray diffraction (XRD). As shown in Fig. 1e, the perovskite films doped with different concentrations of TEAPF₆ show three significant XRD diffraction peaks at 2θ of 14.49°, 28.73°, and 32.15°, corresponding to the (110), (220), and (310) crystal planes of the perovskite film, respectively.^{35–37} The perovskite films with TEAPF₆ do not show any new peaks in the XRD patterns compared to the pristine film, indicating that the TEAPF₆ additive did not change the crystal structure of the perovskite film, which is due to that the larger TEA⁺ cation cannot be

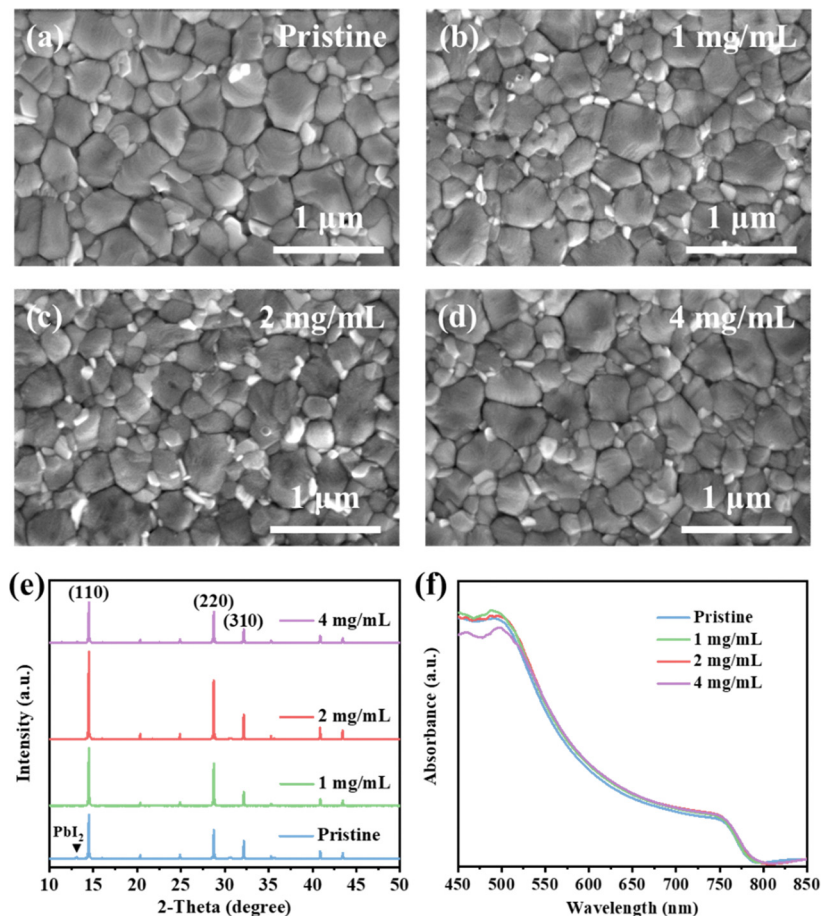


Fig. 1 (a)–(d) SEM images of perovskite films with different concentrations of TEAPF₆. (e) XRD pattern and (f) UV-vis spectra of perovskite films with different concentrations of TEAPF₆.

embedded into the lattice of the perovskite. This phenomenon has also been demonstrated by Yang *et al.*³⁸ For the perovskite films with 1 mg mL⁻¹ and 2 mg mL⁻¹ TEAPF₆, the XRD diffraction peak intensity is enhanced compared to that of the pristine perovskite film, and the XRD peak intensity of the perovskite film with 2 mg mL⁻¹ TEAPF₆ is the highest. Meanwhile, the full width at half maximum (FWHM) of the (110) peak decreases slightly after TEAPF₆ doping, as shown in Fig. S3 (ESI[†]). The increased XRD peak intensity and decreased FWHM indicate that TEAPF₆ doping could improve the crystal quality of the perovskite film, which helps to enhance light absorption, reduce defect density, and reduce charge recombination. Notably, TEAPF₆ doping can suppress the formation of PbI₂ in perovskite films, corresponding to the disappearance of the XRD diffraction peak at a 2θ of 13.08°, which contributes to improving the stability of the perovskite film.⁹ When the doping concentration of TEAPF₆ is increased to 4 mg mL⁻¹, the XRD peak intensity of the perovskite film is decreased and the peak intensity of PbI₂ is increased, indicating that the crystal quality of the perovskite film is decreased.

The crystal quality of the perovskite film affects its optical properties, and we further recorded the ultraviolet-visible (UV-vis) absorption spectra of perovskite films without and

with TEAPF₆. As shown in Fig. 1f, TEAPF₆ doping enhances the light absorption of the perovskite films, which is associated with the improved morphology of the perovskite films after TEAPF₆ doping. The enhanced light absorption is beneficial to improving the short-circuit current of the device. The bandgap of the perovskite can be calculated from UV-vis spectra, as shown in Fig. S4 (ESI[†]). The bandgap (E_g) of the perovskite slightly decreases from 1.58 eV to 1.57 eV after TEAPF₆ doping, which is due to the decreased PbI₂ in the perovskite film after TEAPF₆ doping.

To investigate the charge recombination of the perovskite film without and with TEAPF₆, we obtained the steady-state photoluminescence (PL) and time-resolved PL (TR-PL) spectra. As shown in Fig. 2a, the PL intensity of the perovskite film with 1 mg mL⁻¹ and 2 mg mL⁻¹ TEAPF₆ is increased compared to that of the pristine perovskite film. The increased PL intensity indicates the decreased defects and suppressed charge recombination in perovskite films after TEAPF₆ doping. As the doping concentration of TEAPF₆ is further increased, the PL intensity of perovskite films is decreased, which is due to the deteriorated crystal quality of the perovskite film, corresponding to the XRD results. We also carried out the PL mapping measurements as shown in Fig. 2b and c and Fig. S5 (ESI[†]). The TEAPF₆-doped

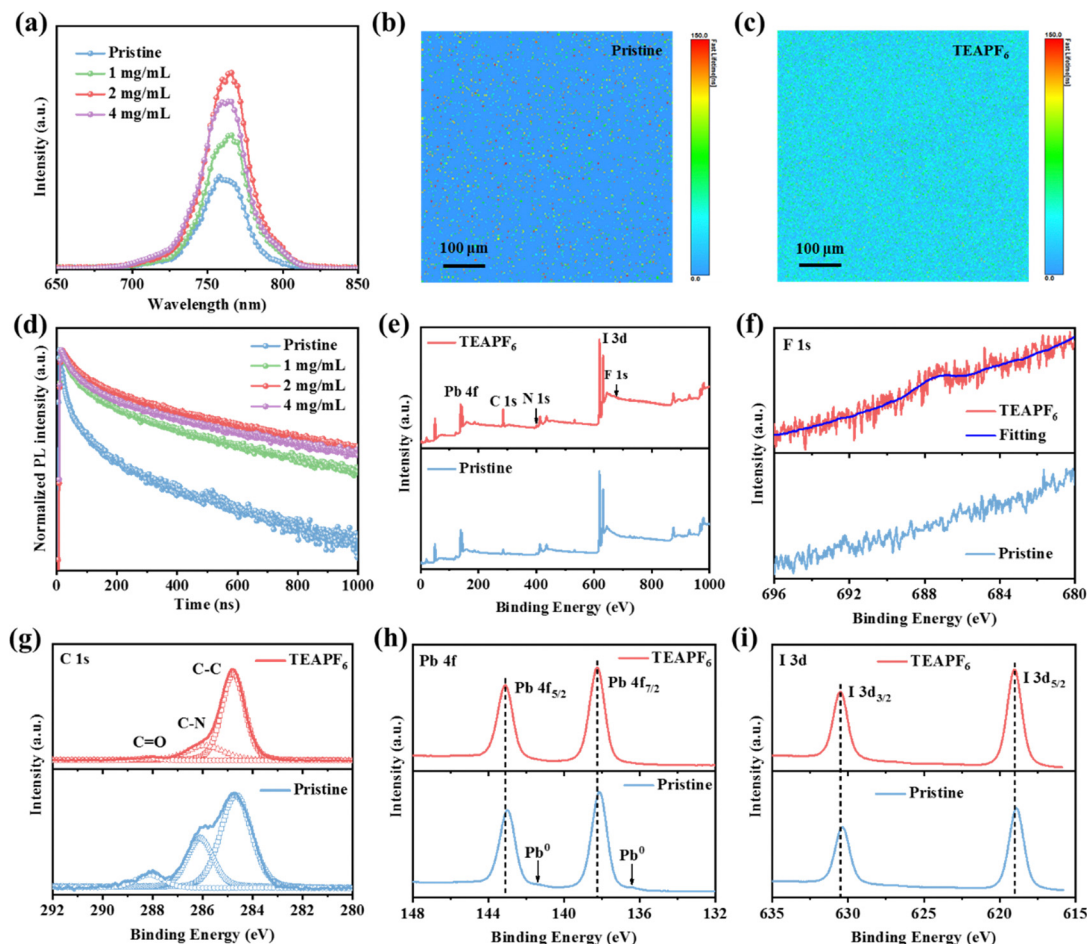


Fig. 2 (a) Steady-state PL spectra of perovskite films. PL mapping of (b) pristine and (c) TEAPF₆ doped perovskite films (2 mg mL⁻¹). (d) TR-PL spectra of perovskite films. (e) The full XPS spectra of perovskite films without and with TEAPF₆. The high-resolution XPS spectra of (f) F 1s, (g) C 1s, (h) Pb 4f, and (i) I 3d of perovskite films.

perovskite film shows higher PL intensity and lifetime compared to the pristine perovskite film, indicating the improved film quality. Fig. 2d shows the TR-PL spectra of perovskite films with different concentrations of TEAPF₆. The curves are fitted by a double exponential decay model as reported in our previous work,^{39,40} and the fitting results are listed in Table S1 (ESI[†]). The pristine perovskite film exhibits a carrier lifetime of 52.95 ns. After TEAPF₆ doping, the carrier lifetime increases to 109.68 ns, 176.87 ns, and 145.41 ns for the perovskite film with 1 mg mL⁻¹, 2 mg mL⁻¹, and 4 mg mL⁻¹ TEAPF₆ doping, respectively. The TR-PL results of the perovskite films correspond to their PL results, indicating that TEAPF₆ doping can suppress the charge recombination in perovskite films caused by defects.

To clarify the interaction between TEAPF₆ and the perovskite, the X-ray photoelectron spectra (XPS) of perovskite films without and with TEAPF₆ are obtained. Fig. 2e and f show the full XPS spectra and F 1s spectra of perovskite films without and with TEAPF₆, respectively. The presence of the F 1s peak indicates the successful incorporation of TEAPF₆ into the perovskite film. As shown in Fig. 2g, the C 1s spectra contain

three peaks corresponding to the C-C, C-N, and C=O peaks. Among them, the C=O peak is considered to be related to the decomposition of organic cations in the perovskite film under the action of oxygen and moisture. The C=O peak of the perovskite film is suppressed after TEAPF₆ doping, indicating that TEAPF₆ doping could improve the stability of the perovskite film.^{17,41} Fig. 2h and i show the Pb 4f and I 3d XPS peaks. The Pb 4f peaks located at 138.13 eV and 143.00 eV shift to higher binding energies of 138.26 eV and 143.13 eV, respectively. The two peaks of I 3d shift from 618.87 eV and 630.38 eV to higher binding energies of 619.00 eV and 630.54 eV, respectively. The shift of Pb 4f and I 3d is because of the strong interaction between TEAPF₆ and the perovskite film. The PF₆⁻ can form a covalent bond with Pb²⁺ and hydrogen bonds with organic cations due to its strong electronegativity. On the other hand, PF₆⁻ can fill halide vacancies in the perovskite film. The strong interaction can suppress the ion migration caused by defects in the perovskite film. It is worth noting that the Pb 4f spectra of the pristine perovskite film exhibit two peaks corresponding to Pb⁰, and the Pb⁰ in the perovskite film affect the optoelectronic characteristics and

stability of the perovskite film.⁴² The Pb^0 peak disappeared in the perovskite film with TEAPF₆, indicating that TEAPF₆ doping could suppress the formation of Pb^0 defects. Furthermore, we recorded the Fourier-transform infrared spectra (FTIR) to confirm the interaction between the PF_6^- and perovskite, as shown in Fig. S6 (ESI†). The TEAPF₆ film exhibits a peak at a wavenumber of 863.49 cm^{-1} , which corresponds to the P-F peak;⁴³ the TEAPF₆-doped perovskite film also shows a P-F peak at 828.77 cm^{-1} , which confirms the presence of PF_6^- in the perovskite film; the shift of P-F stretching vibration mode peaks indicates a strong interaction between the PF_6^- and perovskite. Moreover, the red-shift of the C-H bending vibration (from 1456.98 cm^{-1} to 1443.96 cm^{-1}) is due to the formation of hydrogen bonds caused by the large electronegativity of PF_6^- .⁴⁴

Next, we selected the pristine and 2 mg mL^{-1} TEAPF₆ doped perovskite films to further study the effect of TEAPF₆ doping on the defect density of perovskite films. We measured the defect density of the perovskite film without and with TEAPF₆ by the space charge limited current (SCLC) method. The electron-only device with a structure of ITO/SnO₂/MAFA (TEAPF₆)/PCBM/Ag was fabricated. Fig. S7 (ESI†) shows the dark current-voltage (I - V) curves of devices without and with TEAPF₆. The defect density is defined by the equation: $N_{\text{defect}} = (2\epsilon\epsilon_0 V_{\text{TFL}})/(eL^2)$, where ϵ is the dielectric constant of the perovskite, ϵ_0 is the vacuum dielectric constant, V_{TFL} is the trap-filled limit voltage obtained from I - V curves, e is the electron charge, and L is the

thickness of the perovskite film. The V_{TFL} decreases from 0.93 V to 0.15 V after TEAPF₆ doping, corresponding to the decreased N_{defect} of the perovskite film from $2.68 \times 10^{15}\text{ cm}^{-3}$ to $4.32 \times 10^{14}\text{ cm}^{-3}$. The decreased defect density is due to that TEAPF₆ doping improved the crystal quality and passivated the defects of the perovskite film, which will be conducive to suppressing the charge recombination.

To investigate the effect of TEAPF₆ doping on the PSC performance, we fabricated planar PSCs with a structure of ITO/SnO₂/perovskite (TEAPF₆)/Spiro-OMeTAD/Ag. The details of the fabrication process are shown in the Experimental section in the ESI.† Fig. 3a shows the current density-voltage (J - V) curves of PSCs with different concentrations of TEAPF₆ doping, and the corresponding parameters are listed in Table S2 (ESI†). The pristine PSCs exhibit a champion PCE of 20.23%, with a J_{SC} of 23.89 mA cm^{-2} , a V_{OC} of 1.10 V , and an FF of 0.77. In contrast, the PSCs with 2 mg mL^{-1} TEAPF₆ achieves the highest PCE of 22.13%, with a J_{SC} of 24.26 mA cm^{-2} , a V_{OC} of 1.15 V , and an FF of 0.79. The PSCs with 1 mg mL^{-1} TEAPF₆ exhibit a slightly increased average J_{SC} and FF compared to the pristine PSCs due to the low doping concentration. When the doping concentration of TEAPF₆ is further increased, the average PCE of PSCs is decreased, which is because the high concentration of TEAPF₆ leads to the deterioration of the perovskite film quality and improvement of charge recombination, as proved by XRD and TR-PL. The increase in the PCE is mainly due to a significant increase in V_{OC} and a slight increase

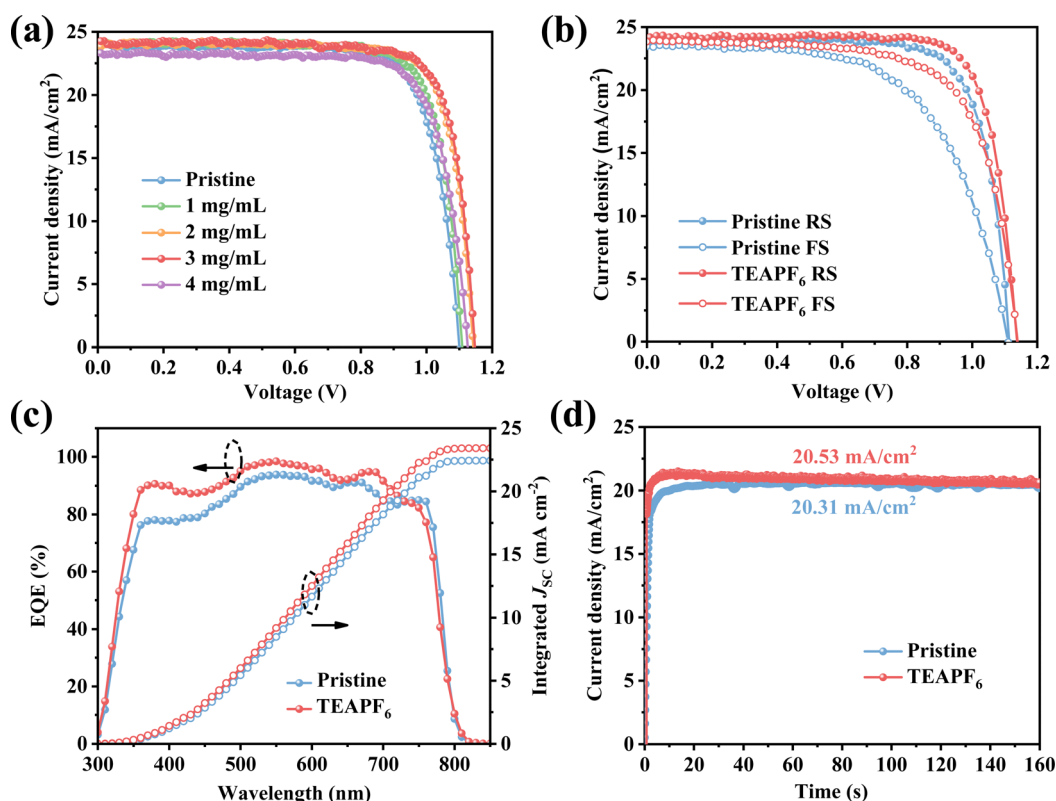


Fig. 3 (a) J - V curves of PSCs with different concentrations of TEAPF₆. (b) J - V curves in different scan directions. (c) EQE spectra and (d) steady-state output J_{SC} of PSCs.

in J_{SC} and FF. The increased J_{SC} and FF are due to the improved crystal quality and reduced grain boundary. The increased V_{OC} is mainly related to passivated defects by TEAPF₆ doping. Fig. 3b shows the forward and reverse J - V curves of the PSCs without and with 2 mg mL⁻¹ TEAPF₆. It is worth noting that hysteresis is suppressed after TEAPF₆ doping. As listed in Table S3 (ESI[†]), the hysteresis index (HI) of PSCs, calculated using $HI = (PCE_{reverse} - PCE_{forward})/PCE_{reverse}$, decreases from 21.65% to 14.23% after TEAPF₆ doping. Research shows that hysteresis is related to defects and ion migration,⁴⁵ and the suppressed hysteresis indicates that TEAPF₆ doping could effectively passivate defects in the perovskite film. To further investigate the effect of TEAPF₆ doping on J_{SC} , the external quantum efficiency (EQE) spectra are recorded. As shown in Fig. 3c, the integrated J_{SC} of PSCs increases from 22.42 mA cm⁻² to 23.41 mA cm⁻² after TEAPF₆ doping, which is consistent with the J - V measurements. The steady-state output of PSCs at maximum power point is also measured, as shown in Fig. 3d.

The device with TEAPF₆ exhibits a steady-state current density of 20.53 mA cm⁻², which is higher than the 20.31 mA cm⁻² of the pristine device. Moreover, the photocurrent of the device with TEAPF₆ shows a shorter stabilization time compared to that of the pristine device, which is consistent with suppressed hysteresis in the TEAPF₆-doped device.

To analyze the effect of TEAPF₆ doping on charge transfer and recombination in the device, we carried out the transient photovoltage (TPV) and transient photocurrent (TPC) measurements. The photovoltage decay lifetime is associated with the charge recombination process in the device.^{46,47} As shown in Fig. 4a, the photovoltage decay time of the device increases from 18.39 μs to 81.05 μs after TEAPF₆ doping, indicating that TEAPF₆ doping can passivate defects and suppress the charge recombination in the device, which will contribute to improving the V_{OC} of the device. The photocurrent decay lifetime is associated with the charge transfer process in the device. As shown in Fig. 4b, the photocurrent decay time of the device

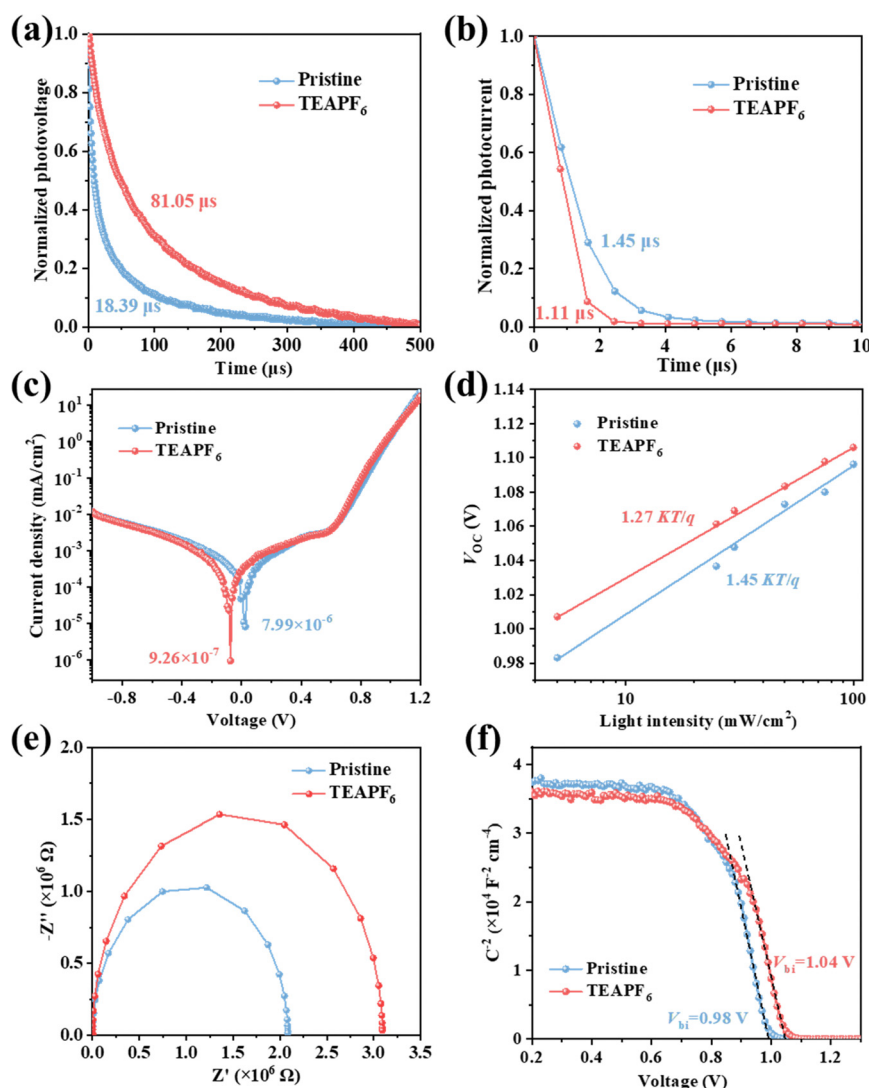


Fig. 4 (a) TPV and (b) TPC measurements, (c) dark J - V measurements, (d) V_{OC} as a function of light intensity, (e) EIS measurements, and (f) C - V measurements of the PSCs without and with TEAPF₆.

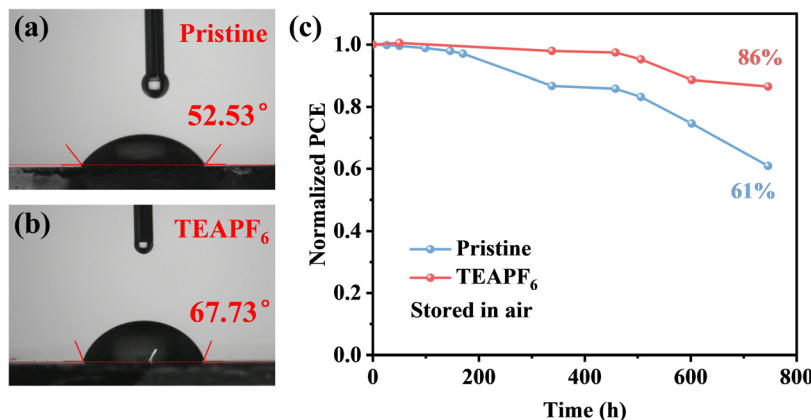


Fig. 5 (a) and (b) The water contact angle of perovskite films without and with TEAPF₆. (c) The long-term stability of PSCs.

decreases from 1.45 μs to 1.11 μs after TEAPF₆ doping, indicating that TEAPF₆ can improve the charge extraction and transfer in the device, which helps to improve the J_{SC} of the device. Fig. 4c shows the dark J - V curves of devices, and the dark current density of the device after TEAPF₆ doping is 1 order of magnitude lower than that of the pristine devices. Dark current is related to charge recombination. The lower dark current further demonstrates the suppressed charge recombination by TEAPF₆ doping. The relationship between V_{OC} and light intensity (P) is also used to clarify the charge recombination process in devices. As shown in Fig. 4d, V_{OC} and P exhibit a linear relationship that obeys the equation: $V_{\text{OC}} = nKT/q \ln(P)$, where K is the Boltzmann constant, T is the absolute temperature, and q is the elementary charge, respectively. The slope of $V_{\text{OC}}-P$ curves (nKT/q) reflects the nonradiative recombination in the device.⁴⁸ The slope of the device with TEAPF₆ ($1.27 KT/q$) is much lower than that of the undoped device ($1.45 KT/q$), indicating that TEAPF₆ doping can suppress the nonradiative recombination in the device. The electrical impedance spectra (EIS) are also recorded to investigate the charge transfer and recombination in the device. Fig. 4e shows the Nyquist plots of the device without and with TEAPF₆; the curves are fitted by the equivalent circuit shown in Fig. S8 (ESI[†]), and the fitting parameters are listed in Table S4 (ESI[†]). The pristine device shows a series resistance (R_{s}) of 60.93 Ω and a recombination resistance of (R_{rec}) of $2.08 \times 10^6 \Omega$, and the device with TEAPF₆ shows a R_{s} of 41.48 Ω and a R_{rec} of $3.09 \times 10^6 \Omega$. The increased R_{rec} indicates that TEAPF₆ doping effectively suppresses charge recombination in PSCs. We further carried out the capacitance-voltage (C - V) measurement to clarify the effect of TEAPF₆ doping on built-in potential (V_{bi}). The relationship between capacitance and V_{bi} obeys the Mott-Schottky formula.^{17,49} As shown in Fig. 4f, the V_{bi} of the device increases from 0.98 V to 1.04 V after TEAPF₆ doping. The increased V_{bi} of the device after TEAPF₆ doping is due to the suppressed charge recombination and helps to improve the charge extraction and transfer in the device.

The stability of PSCs is critical for their commercial application. The stability of PSCs is not only affected by the properties of the

perovskite itself but also related to external factors such as moisture, oxygen, and heat. We have demonstrated that TEAPF₆ can improve the crystal quality of perovskite films and passivate defects. The pseudohalide PF₆⁻ can bond with Pb²⁺ and organic cations and fill the halide vacancies, which can enhance the crystal structure of perovskite films. Moreover, the larger TEA⁺ cation cannot be embedded into the lattice of the perovskite but can be absorbed onto the perovskite surface by a hydrogen bond, thus the hydrophobic TEA⁺ and PF₆⁻ help to improve the stability of the perovskite in moisture. First, we measured the water contact angle of perovskite films without and with TEAPF₆. As shown in Fig. 5a and b, the water contact angle of the perovskite films is improved from 52.53° to 67.73° after TEAPF₆ doping. The improved hydrophobicity prevents perovskite decomposition caused by water in the environment. Then, the PSCs without encapsulation are stored in an ambient environment with a relative humidity of 40% at room temperature in the dark. As shown in Fig. 5c, the pristine device retains 61% of its initial PCE after 750 h of storage, and the device with TEAPF₆ retains a remarkable 86% of its initial PCE under the same conditions. The increased long-term stability of PSCs is due to the improved crystal quality and hydrophobicity of the perovskite film after TEAPF₆ doping.

3. Conclusion

In conclusion, we have introduced the multifunctional IL TEAPF₆ as an additive into the perovskite film. The pseudohalide PF₆⁻ can enhance the crystal structure and passivate the defects in the perovskite film by bonding with uncoordinated Pb²⁺ and the organic cation and filling the halide vacancies, which helps to suppress the ion migration and charge recombination in the perovskite film. Meanwhile, the TEAPF₆ doping can also improve the hydrophobicity of the perovskite film. As a result, the PSCs with TEAPF₆ achieve a PCE of 22.13%, with a significantly increased V_{OC} from 1.10 V to 1.15 V. Remarkably, TEAPF₆ doping enhances the stability of devices, and the unencapsulated device with TEAPF₆ retains 86% of its initial

PCE after aging in an ambient environment with a relative humidity of 40% for 750 h. This work provides an excellent strategy based on multifunctional pseudohalide-based ionic liquid doping to improve the efficiency and stability of PSCs.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was financially supported by the National Natural Science Foundation of China (52192610, 62274127), the National Key Research and Development Program of China (2021YFA0715600, 2021YFA0717700, 2018YFB2202900), the Natural Science Basic Research Program of Shaanxi (2023-JC-QN-0681), Wuhu and Xidian University special fund for industry-university-research cooperation (XWYCX-012021004), and Fundamental Research Funds for the Central Universities (XJS222214).

References

- J. H. Noh, S. H. Im, J. H. Heo, T. N. Mandal and S. Il Seok, *Nano Lett.*, 2013, **13**, 1764–1769.
- Q. Dong, Y. Fang, Y. Shao, P. Mulligan, J. Qiu, L. Cao and J. Huang, *Science*, 2015, **347**, 967–970.
- L. Zhou, J. Chang, Z. Liu, X. Sun, Z. Lin, D. Chen, C. Zhang, J. Zhang and Y. Hao, *Nanoscale*, 2018, **10**, 3053–3059.
- Z. Liu, J. Chang, Z. Lin, L. Zhou, Z. Yang, D. Chen, C. Zhang, S. (Frank) Liu and Y. Hao, *Adv. Energy Mater.*, 2018, **8**, 1703432.
- J. He, Q. Wang, Y. Xu, X. Guo, L. Zhou, J. Su, Z. Lin, J. Zhang, Y. Hao and J. Chang, *Small*, 2023, **19**, 2205962.
- A. Kojima, K. Teshima, Y. Shirai and T. Miyasaka, *J. Am. Chem. Soc.*, 2009, **131**, 6050–6051.
- J. Burschka, N. Pellet, S.-J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin and M. Grätzel, *Nature*, 2013, **499**, 316–319.
- W. S. Yang, B. Park, E. H. Jung, N. J. Jeon, Y. C. Kim, D. U. Lee, S. S. Shin, J. Seo, E. K. Kim, J. H. Noh and S. Il Seok, *Science*, 2017, **356**, 1376–1379.
- Y. Zhao, F. Ma, Z. Qu, S. Yu, T. Shen, H. Deng, X. Chu, X. Peng, Y. Yuan, X. Zhang and J. You, *Science*, 2022, **377**, 531–534.
- M. Jeong, I. W. Choi, E. M. Go, Y. Cho, M. Kim, B. Lee, S. Jeong, Y. Jo, H. W. Choi, J. Lee, J.-H. Bae, S. K. Kwak, D. S. Kim and C. Yang, *Science*, 2020, **369**, 1615–1620.
- NREL, Best Research-Cell Efficiencies, <https://www.nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.pdf>.
- J. Park, J. Kim, H.-S. Yun, M. J. Paik, E. Noh, H. J. Mun, M. G. Kim, T. J. Shin and S. Il Seok, *Nature*, 2023, 1–3.
- S. Rühle, *Sol. Energy*, 2016, **130**, 139–147.
- J. J. Yoo, G. Seo, M. R. Chua, T. G. Park, Y. Lu, F. Rotermund, Y.-K. Kim, C. S. Moon, N. J. Jeon, J.-P. Correa-Baena, V. Bulović, S. S. Shin, M. G. Bawendi and J. Seo, *Nature*, 2021, **590**, 587–593.
- L. Zhou, Z. Lin, Z. Ning, T. Li, X. Guo, J. Ma, J. Su, C. Zhang, J. Zhang, S. Liu, J. Chang and Y. Hao, *Sol. RRL*, 2019, **3**, 1900293.
- J. Ma, Z. Lin, X. Guo, L. Zhou, J. He, Z. Yang, J. Zhang, Y. Hao, S. Liu and J. Chang, *J. Energy Chem.*, 2021, **63**, 558–565.
- X. Guo, X. Huang, J. Su, Z. Lin, J. Ma, J. Chang and Y. Hao, *Chem. Eng. J.*, 2021, **417**, 129184.
- F. Gao, Y. Zhao, X. Zhang and J. You, *Adv. Energy Mater.*, 2020, **10**, 1902650.
- R. Fan, W. Zhou, Z. Huang and H. Zhou, *EnergyChem*, 2020, **2**, 100032.
- T. Niu, L. Chao, W. Gao, C. Ran, L. Song, Y. Chen, L. Fu and W. Huang, *ACS Energy Lett.*, 2021, **6**, 1453–1479.
- M. Shahiduzzaman, E. Y. Muslih, A. K. M. Hasan, L. Wang, S. Fukaya, M. Nakano, M. Karakawa, K. Takahashi, M. Akhtaruzzaman, J.-M. Nunzi and T. Taima, *Chem. Eng. J.*, 2021, **411**, 128461.
- L. Chao, T. Niu, Y. Xia, Y. Chen and W. Huang, *Acc. Mater. Res.*, 2021, **2**, 1059–1070.
- M. Li, C. Zhao, Z. Wang, C. Zhang, H. K. H. Lee, A. Pockett, J. Barbé, W. C. Tsoi, Y. Yang, M. J. Carnie, X. Gao, W. Yang, J. R. Durrant, L. Liao and S. M. Jain, *Adv. Energy Mater.*, 2018, **8**, 1801509.
- D. He, T. Zhou, B. Liu, L. Bai, W. Wang, H. Yuan, C. Xu, Q. Song, D. Lee, Z. Zang, L. Ding and J. Chen, *EcoMat*, 2022, **4**, 1–14.
- X. Zhu, M. Du, J. Feng, H. Wang, Z. Xu, L. Wang, S. Zuo, C. Wang, Z. Wang, C. Zhang, X. Ren, S. Priya, D. Yang and S. (Frank) Liu, *Angew. Chem., Int. Ed.*, 2021, **60**, 4238–4244.
- Y. Wang, Y. Yang, N. Li, M. Hu, S. R. Raga, Y. Jiang, C. Wang, X. Zhang, M. Lira-Cantu, F. Huang, Y. Cheng and J. Lu, *Adv. Funct. Mater.*, 2022, **32**, 2204396.
- Y. Du, Q. Tian, X. Chang, J. Fang, X. Gu, X. He, X. Ren, K. Zhao and S. (Frank) Liu, *Adv. Mater.*, 2022, **34**, 2106750.
- S. Akin, E. Akman and S. Sonmezoglu, *Adv. Funct. Mater.*, 2020, **30**, 2002964.
- N. Li, S. Tao, Y. Chen, X. Niu, C. K. Onwudinanti, C. Hu, Z. Qiu, Z. Xu, G. Zheng, L. Wang, Y. Zhang, L. Li, H. Liu, Y. Lun, J. Hong, X. Wang, Y. Liu, H. Xie, Y. Gao, Y. Bai, S. Yang, G. Brocks, Q. Chen and H. Zhou, *Nat. Energy*, 2019, **4**, 408–415.
- L. Chu, *Matter*, 2021, **4**, 1762–1764.
- J. Tao, X. Liu, J. Shen, S. Han, L. Guan, G. Fu, D. Kuang and S. Yang, *ACS Nano*, 2022, **16**, 10798–10810.
- H. Cheng, C. Liu, J. Zhuang, J. Cao, T. Wang, W. Wong and F. Yan, *Adv. Funct. Mater.*, 2022, **32**, 2204880.
- S. Bai, P. Da, C. Li, Z. Wang, Z. Yuan, F. Fu, M. Kawecki, X. Liu, N. Sakai, J. T.-W. Wang, S. Huettner, S. Buecheler, M. Fahlman, F. Gao and H. J. Snaith, *Nature*, 2019, **571**, 245–250.
- Y. Lin, N. Sakai, P. Da, J. Wu, H. C. Sansom, A. J. Ramadan, S. Mahesh, J. Liu, R. D. J. Oliver, J. Lim, L. Aspirtarte, K. Sharma, P. K. Madhu, A. B. Morales-Vilches, P. K. Nayak,

- S. Bai, F. Gao, C. R. M. Grovenor, M. B. Johnston, J. G. Labram, J. R. Durrant, J. M. Ball, B. Wenger, B. Stannowski and H. J. Snaith, *Science*, 2020, **369**, 96–102.
- 35 B. Zhang, J. Su, X. Guo, L. Zhou, Z. Lin, L. Feng, J. Zhang, J. Chang and Y. Hao, *Adv. Sci.*, 2020, **7**, 1903044.
- 36 X. Guo, J. Du, Z. Lin, J. Su, L. Feng, J. Zhang, Y. Hao and J. Chang, *Chem. Eng. J.*, 2021, **407**, 127997.
- 37 X. Guo, X. Huang, J. Su, Z. Lin, J. Ma, J. Chang and Y. Hao, *Chem. Eng. J.*, 2021, **417**, 129184.
- 38 S. Yang, Y. Wang, P. Liu, Y.-B. Cheng, H. J. Zhao and H. G. Yang, *Nat. Energy*, 2016, **1**, 15016.
- 39 B. Zhang, J. Su, X. Guo, L. Zhou, Z. Lin, L. Feng, J. Zhang, J. Chang and Y. Hao, *Adv. Sci.*, 2020, **7**, 1903044.
- 40 L. Zhou, X. Guo, Z. Lin, J. Ma, J. Su, Z. Hu, C. Zhang, S. (Frank) Liu, J. Chang and Y. Hao, *Nano Energy*, 2019, **60**, 583–590.
- 41 L. Zhou, J. Su, Z. Lin, X. Guo, J. Ma, T. Li, J. Zhang, J. Chang and Y. Hao, *Research*, 2021, **2021**, 1–11.
- 42 L. Wang, H. Zhou, J. Hu, B. Huang, M. Sun, B. Dong, G. Zheng, Y. Huang, Y. Chen, L. Li, Z. Xu, N. Li, Z. Liu, Q. Chen, L.-D. Sun and C.-H. Yan, *Science*, 2019, **363**, 265–270.
- 43 J. Tao, X. Liu, J. Shen, S. Han, L. Guan, G. Fu, D.-B. Kuang and S. Yang, *ACS Nano*, 2022, **16**, 10798–10810.
- 44 Z. Li, Z. Li, G. Peng, C. Shi, H. Wang, S. Ding, Q. Wang, Z. Liu and Z. Jin, *Adv. Mater.*, 2023, 2300480.
- 45 C. Li, S. Tscheuschner, F. Paulus, P. E. Hopkinson, J. Kießling, A. Köhler, Y. Vaynzof and S. Huettner, *Adv. Mater.*, 2016, **28**, 2446–2454.
- 46 J. He, J. Su, J. Di, Z. Lin, S. Zhang, J. Ma, J. Zhang, S. Liu, J. Chang and Y. Hao, *Nano Energy*, 2022, **94**, 106960.
- 47 X. Guo, J. Su, Z. Lin, X. Wang, Q. Wang, Z. Zeng, J. Chang and Y. Hao, *iScience*, 2021, **24**, 102276.
- 48 Q. Jiang, Y. Zhao, X. Zhang, X. Yang, Y. Chen, Z. Chu, Q. Ye, X. Li, Z. Yin and J. You, *Nat. Photonics*, 2019, **13**, 460–466.
- 49 L. Zhou, J. Su, Z. Lin, X. Guo, J. Ma, T. Li, J. Zhang, J. Chang and Y. Hao, *Research*, 2021, **2021**, 1–11.