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## Efficient monolithic perovskite–Si tandem solar cells enabled by an ultra-thin indium tin oxide interlayer†

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Tandem solar cells are promising for overcoming the theoretical single junction solar cell efficiency limit. Here we report an elegant approach to monolithically integrate perovskites and Si-heterojunction tandem solar cells using an ultra-thin indium tin oxide (ITO) interlayer. It is found that a 1.7 nm ITO interlayer is sufficient to provide the interfacing between the silicon and perovskite subcells with minimum parasitic resistance and efficient trap states for carrier recombination. Thinner ITO interlayers result in discontinuous films while thicker ITO interlayers with improved lateral conductivity are less effective in localising shunts especially for large ( $\geq 1$  cm<sup>2</sup>) area tandem cells. Furthermore, it is found that a tandem cell with a 26 nm thick (p)a-Si:H junction in the bottom cell performs better than that with a 13 nm thick junction. When combined with a 1.7 nm ultra-thin ITO layer and a front anti-reflection coating, our champion 1.0 cm<sup>2</sup> cell achieved 27.2% power conversion efficiency (PCE) and a fill factor (FF) of 82.4% under reverse scan. This FF is the highest of any perovskite–Si tandem cell. The ultra-thin ITO layer is suitable for large area cells. 24.2% efficiency was demonstrated on an 11.8 cm<sup>2</sup> cell. A 21.1% efficient 65.1 cm<sup>2</sup> (~4 inch round) cell was also fabricated. This representative cell shows negligible efficiency loss after 600 hours of continuous 1 sun illumination under MPP tracking. This work provides insights into the role of ultra-thin interfacing ITO layers for monolithic large-area perovskite–Si tandem cells.

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### Broader context

According to the most recent industry roadmap ITRPV (2022), Si-based tandem solar cells will become part of the photovoltaic technology mix starting in 2026. Perovskite solar cell technology has experienced a very rapid rate of improvement in energy conversion efficiency amongst all solar technologies developed to date. The technology has the performance credentials and the ease of fabrication to be a candidate for tandem solar cells. Perovskite–Si tandem cell research has an attracted immense amount of interest in the last 5 years and has experienced an even faster rate of improvement in energy conversion efficiency. This work reports that ultra-thin indium tin oxide (ITO) layers as thin as 1.7 nm can be used to integrate perovskite and silicon solar cells for monolithic tandem cells thereby reducing the associated manufacturing cost and optical loss. While such a thin layer is sufficient for carrier transport between the Si cell front junction and the perovskite cell rear contact, it has limited lateral conductivity which is an advantage for larger area perovskite–Si tandem cells that are prone to shunting. In fact, this principle can be applied to any kind of tandem cell that has “imperfect” charge selective layers prone to shunting regardless of the polarity or material choice and is most useful for scaling up to large areas.

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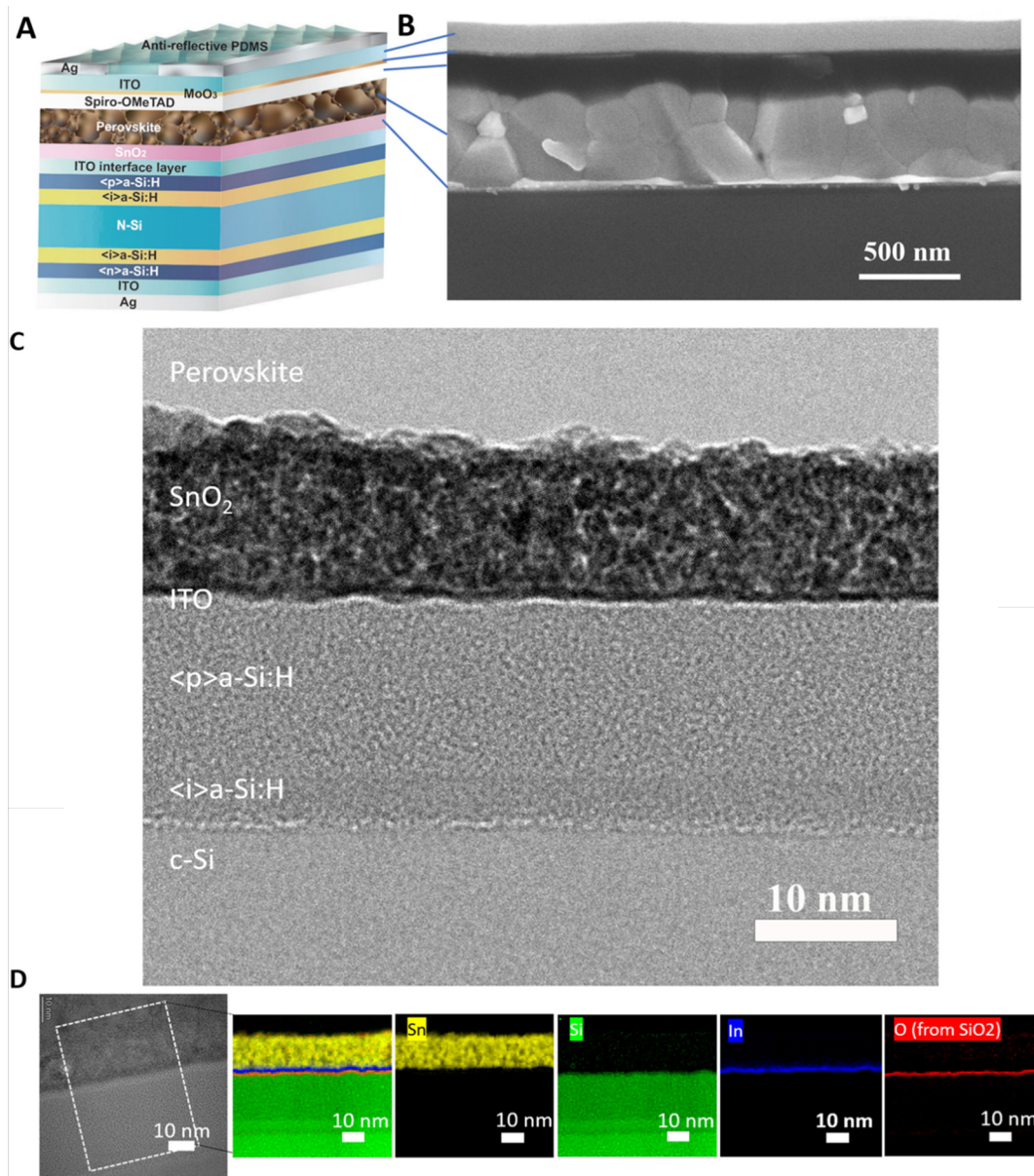
Crystalline silicon (c-Si) photovoltaic technology dominates the commercial market due to its low cost and reliability.<sup>1</sup> The power conversion efficiency (PCE) of the best laboratory c-Si cell has reached 26.7%,<sup>2</sup> approaching a single junction theoretical limit of 29.4%.<sup>3</sup> Perovskite photovoltaic cells with superior optical absorption and ease of fabrication have attracted strong research and development activity, resulting in a rapid improvement in the PCE, reaching 25.7% (independently certified), from 14.1%, in less than 10 years.<sup>4</sup> One of the most promising ways to commercialize perovskite solar cells and boost the efficiency of the incumbent technology is to integrate perovskites and Si solar cells into double junction tandem cells that overcome the theoretical efficiency limit of a single junction cell.<sup>5</sup> A 2-terminal perovskite–Si tandem device in which cells are monolithically integrated represents the most elegant way of reducing wiring complexities. The first monolithic perovskite–Si tandem device was reported in 2015 with an efficiency of 13.6%.<sup>6</sup> Recently, independently certified efficiencies of 31.5%<sup>4</sup> and 32.5% on  $\sim 1 \text{ cm}^2$  have been reported in a press release by the research groups in École Polytechnique Fédérale de Lausanne (EPFL)/Centre Suisse d'Electronique et de Microtechnique (CSEM) and Helmholtz-Zentrum Berlin (HZB), respectively. Table S1 (ESI<sup>†</sup>) summarizes the 2-terminal perovskite–Si tandem solar cells reported to date. Most of these tandem cells are demonstrated on small areas ( $\sim 1 \text{ cm}^2$ ). For a  $>10 \text{ cm}^2$  cell demonstration reported in peer-reviewed literature, Sahli *et al.* employed a Si heterojunction (SHJ) cell as the bottom cell and achieved a steady-state PCE of 18.0% on  $13 \text{ cm}^2$  in 2017.<sup>7</sup> Our previous work employed Si homo-junction cells as the bottom cells and achieved a steady-state PCE of 21.8% on  $16 \text{ cm}^2$ .<sup>8</sup> Following that, Kamino *et al.* demonstrated a  $57.4 \text{ cm}^2$  tandem cell with a steady-state PCE of 22.6% using an SHJ cell as the bottom cell.<sup>9</sup> This is the highest efficiency for the largest area tandem peer reviewed reported until now. The same group also fabricated a 24.3%  $57.4 \text{ cm}^2$  cell as per a technical report.<sup>10</sup> Most recently, an independently certified efficiency of 26.8% on a 6 inch full size wafer ( $274.22 \text{ cm}^2$ ) has been announced by Oxford PV.<sup>11</sup>

The interfacing between the top and bottom cells in a tandem cell is key to high-performance. An indium tin oxide (ITO) layer of 20 to 120 nm is the most commonly used interface layer<sup>12–33</sup> for perovskite–Si tandem cells. As the ITO interface layer thickness increases, so does parasitic optical loss. Another advantage of thin (or even no) ITO interface layer is its poor lateral conductivity which is desirable for localising shunting effects especially important for large area n–i–p perovskite–Si tandem cells that rely on the use of the solution processed  $\text{SnO}_2$  layer as the electron transport layer (ETL). They (see green entries in Table S1, ESI<sup>†</sup>) are more prone to shunting which worsens with device area. For these devices that use a thick ITO interlayer (80 nm),  $\sim 80\%$  FF could only be achieved if the device areas were small, e.g.,  $0.03 \text{ cm}^2$  for a champion device FF of 79%<sup>16</sup> and  $0.06 \text{ cm}^2$  for a champion device FF of 81%.<sup>19</sup> Interestingly, in the study in ref. 19, as the device area was increased to  $0.16 \text{ cm}^2$ , the ITO interlayer was thinner at 10 nm for a FF of 81% to be achieved. As device areas increased, e.g., to  $4 \text{ cm}^2$  as shown in the study in ref. 41 and ref. 34 and

$16 \text{ cm}^2$  as in ref.8, a FF of 78% to 81% could only be achieved when the ITO layer = 0 nm for n–i–p perovskite–Si tandem cells that use solution processed  $\text{SnO}_2$  layers.

In this work, we report an efficient monolithic perovskite Si-heterojunction tandem solar cell enabled by an ultra-thin indium tin oxide (ITO) interlayer. It is found that a 1.7 nm ITO interlayer is sufficient to provide the interfacing between for the silicon and perovskite subcells. Compared with the tandem device without ITO, the electronic capture from the  $\text{SnO}_2$  side has been enhanced due to the lower valence band of the ultra-thin ITO layer and reduced valence band offset between  $\text{SnO}_2$  and  $\langle p \rangle$ a-Si:H. When this ultra-thin ITO layer is applied to the perovskite–Si tandem cell, a fill factor (FF) of 82.4% was achieved by the champion 27.2% efficient  $1.0 \text{ cm}^2$  cell under reverse scan. This FF is the highest for any perovskite–Si tandem cell demonstrated so far. Such tandem cell design using ultra-thin ITO interlayers is also demonstrated on large areas achieving reverse scan efficiencies of 24.2% and 21.1% on  $11.8 \text{ cm}^2$ , and  $65.1 \text{ cm}^2$ , respectively. Device stability has also been demonstrated by a representative cell, showing negligible efficiency loss after 600 hours of operation at its maximum power point under continuous 1 sun illumination.

Fig. 1A shows the schematic of the developed perovskite–Si tandem solar cell structure. The bottom cell consists of a Si heterojunction cell with hydrogenated amorphous silicon (a-Si:H) passivation layers on both sides which has been demonstrated to produce a high open circuit voltage ( $V_{\text{OC}}$ ) consistently. The top cell consists of a  $\text{SnO}_2/(\text{FAPbI}_3)_{0.83}(\text{MAPbBr}_3)_{0.17}/2,2',7,7'$ -tetrakis(*N,N*-di-*p*-methoxyphenylamine)-9,9-spirobifluorene (Spiro-OMeTAD)/ $\text{MoO}_3$ /ITO/Ag grid with an optically anti-reflective polydimethylsiloxane (PDMS) layer. Fig. 1B shows the SEM cross-sectional image of a typical perovskite top cell in the tandem cell stack. ( $\text{SnO}_2 \sim 15 \text{ nm}$ , perovskite layer  $\sim 480 \text{ nm}$ , spiro-OMeTAD  $\sim 150 \text{ nm}$ ,  $\text{MoO}_3 \sim 10 \text{ nm}$ , and top ITO electrode  $\sim 135 \text{ nm}$ ). For the interlayer of the tandem device, we proceeded to fabricate ultra-thin ITO  $\leq 5 \text{ nm}$  by a sputter deposition process that gave excellent thickness control. The thicknesses of the ITO inter-layers were confirmed by fitting their measured ellipsometric data (Fig. S1, ESI<sup>†</sup>). To examine a representative  $\text{SnO}_2/1.7 \text{ nm ITO}/26 \text{ nm } \langle p \rangle$ a-Si:H interface stack (Fig. 1C), we carried out cross-sectional scanning transmission electron microscopy (STEM) and electron energy loss spectroscopy (EELS) for elemental mapping. Results are shown in Fig. 1D and Fig. S2 (ESI<sup>†</sup>) respectively. While energy dispersive spectroscopy (EDS) can be used to map the elements Sn, Si, In and O, EELS allows us to resolve the stack into the  $\text{SnO}_2$ , ITO, native  $\text{SiO}_2$  and  $\langle p \rangle$ a-Si:H layers which will be more challenging in EDS. For example, the In-L $\delta$  and Sn-L $\delta, \beta$  EDS signals from ITO and  $\text{SnO}_2$  layers, respectively, are very close to each other making it hard to distinguish between the layers. This can also result in an overestimation of the ultra-thin ITO layer thickness if the In-L $\delta$  signal “overshadows” the Sn-L $\delta, \beta$  signal. EELS on the other hand allows a much more accurate determination of the thickness of each layer. Details on how this is done can be found in the caption of Fig. S2 (ESI<sup>†</sup>). The thicknesses of the  $\text{SnO}_2$ /ITO coupling stack on the  $\langle p \rangle$ a-Si:H/ $\langle i \rangle$ a-Si:H passivating junction,



**Fig. 1** Schematic illustration of the monolithic perovskite/Si tandem solar cell and characterisation of the  $\text{SnO}_2$ /ultra-thin ITO interface. (A) Schematic of the perovskite–silicon tandem cell structure. (B) Cross-sectional scanning electron microscopy (SEM) image of a typical perovskite top cell in the tandem device. (C) Bright field transmission electron microscopy (TEM), showing the interfacial layers with different crystallinities, including the underlying crystalline Si wafer surface. (D) Cross-sectional scanning transmission electron microscopy (STEM) and the corresponding electron energy loss spectroscopy (EELS) maps of Sn, Si, In and O, allowing the  $\text{SnO}_2$ /ITO/native  $\text{SiO}_2$ / $\langle p \rangle$ a-Si:H structure to be resolved (spectrum image of this region and the corresponding EELS spectra can be found in Fig. S2, ESI<sup>†</sup>).

determined from EELS, are 15 nm for  $\text{SnO}_2$ , 1.7 nm for ITO, 26 nm for  $\langle p \rangle$ a-Si:H and 6 nm for  $\langle i \rangle$ a-Si:H. They matched those predicted by the deposition processes. A  $\sim 1.5$  nm native  $\text{SiO}_2$  layer was also observed on the Si wafer surface. To ascertain the uniformity of the 1.7 nm ITO inter-layer over a large area, we carried out high-angle annular dark-field (HAADF) STEM imaging on 3 different points near the edge of a  $65.1 \text{ cm}^2$  ( $\sim 4$  inch round) tandem cell. Results are shown in Fig. S3A–C (ESI<sup>†</sup>). We also carried out energy dispersive spectroscopy (EDS) mappings of cross-sectional STEM images taken from three points (middle, edge and a point in between) across a 4 inch round tandem cell.

Results are shown in from the edge (position: point 3 as shown in Fig. S3, ESI<sup>†</sup>). Uniform ultra-thin ITO distribution can be directly observed from HAADF images. Results are shown in Fig. S3D–S3F (ESI<sup>†</sup>). All of the results above as well as results of conductive atomic force microscopy (c-AFM) (Fig. S4c, ESI<sup>†</sup>) indicated that the 1.7 nm ITO layer is continuous, similar to that shown in Fig. 1. However, as the thickness of the ITO layer is further reduced to 1 nm, the layer is no longer continuous (Fig. S4b, ESI<sup>†</sup>). “Islanding” can be observed in the 1.0 nm ITO layer by cross-sectional STEM especially at a tilt (Fig. S5b, ESI<sup>†</sup>).

To investigate the effect of the ITO thickness on the tandem cell performance, we fixed the thickness of the (p)a-Si:H in the Si bottom cells at 13 nm<sup>35</sup> and we fabricated 1 cm<sup>2</sup> tandem test cells with ultra-thin ( $\leq 5$  nm) ITO layers at 5 different thicknesses. These test cells have identical structures as shown in

Fig. 1A except that they did not have antireflective PDMS layers. The  $J$ - $V$  curves of the champion testing tandem devices with different ITO interlayer thickness are shown in Fig. 2A. Their respective Suns- $V_{OC}$  curves are shown in Fig. S6 (ESI†) to determine the trend of series resistance ( $R_s$ ) and pseudo-FF



**Fig. 2** Effect of the ITO inter-layer thickness on the performance of 1 cm<sup>2</sup> tandem test solar cells before the application of front anti-reflection control. (A)  $J$ - $V$  curve of the champion testing tandem devices with different ITO interlayer thicknesses. Distribution of (B) PCE, (C) short-circuit current density ( $J_{sc}$ ), (D) fill factor (FF), and (E) open circuit voltage ( $V_{oc}$ ) based on 10 devices for each condition. The highest value is the maximum value. The highest bar is the 75<sup>th</sup> percentile value. The middle bar is the median value. The square mark is for the average. The lowest bar is the 25<sup>th</sup> percentile value. The lowest value is the minimum.

(pFF) of tandem solar cells with varying interlayer ITO thicknesses (Table 1). The distribution of the electrical characteristics of the test cells is shown in Fig. 2B–D and is summarised in Table 1.  $J_{SC}$  remained constant as the ITO thickness varied, in agreement with optical simulation (Fig. S7, ESI†).  $V_{OC}$  and especially FF were the highest when the ITO was 1.7 nm thick, maximizing PCE but decreased when the ITO thickness was increased or reduced further (*e.g.*, >1.7 nm or <1.7 nm). With respect to  $R_S$ , the devices followed a similar trend to the FF while the  $R_{SH}$  decreased with the ITO thickness (Fig. S8, ESI†).

The decrease in  $R_{SH}$  with ITO thickness was due to the improved lateral conductivity in the ITO layers failing to localise shunting effects. The improved lateral conductivity was confirmed by sheet resistance measurements (Fig. S9A, ESI†) and the evidence of increased shunting could be seen by the appearance of the humps in the lower voltage region of the dark current–voltage (*IV*) characteristics of the tandem devices with 3.5 nm and 5 nm ITO as shown in Fig. S10 (ESI†). The number of shunted and low- $V_{OC}$  cells also increase with 3.5 nm and 5 nm ITO (Fig. S11, ESI†) thereby lowering the mean  $V_{OC}$  values observed in Fig. 2 and Table 1. It is important to note that the function of ultra-thin ITO localising shunts is effective in cells with areas  $\geq 1 \text{ cm}^2$  (*i.e.*, commercially relevant cells) and is not obvious in small cells. To verify this, we fabricated small ( $0.09 \text{ cm}^2$ ) perovskite–Si tandem devices with 5 nm or 20 nm ITO interlayers. The results in Fig. S12 (ESI†) show much less sensitivity to ITO interlayer thickness and higher efficiencies (23.5% for the champion cell with a 5 nm ITO interlayer and 23.0% for the champion cell with a 20 nm ITO interlayer) could be achieved due to shunting being less severe but only on a very small area *e.g.*,  $0.09 \text{ cm}^2$ .

In terms of  $R_S$ , it is at its minimum in the case of 1.7 nm ITO layer, as the continuous film had sufficient conductivity for carrier transport in the vertical direction (Fig. S9B, ESI†). For both 1 nm and 0 nm ITO cases,  $R_S$  was higher and an “S-shape” could be observed in the light *JV* curves (Fig. 2A). The higher  $R_S$  for the 1 nm ITO case can be explained by the discontinuous film (Fig. S4 and S5, ESI†). To investigate the cause of the “S-shape”, Sentaurus technology computer-aided design (TCAD) was used to model the energy band structure of the tandem cells under thermal equilibrium with and without ITO. The results are shown in Fig. 3. In the presence of a continuous ITO film as in the case of 1.7 nm ITO, due to the high trap density of ITO, the ultra-thin layers provide sufficient trap states for the interface between the ITO layer and the (p)a-Si:H layer *via* trap assisted tunnelling.<sup>36,37</sup> In the case of 1 nm and 0 nm ITO

layers, the  $\text{SnO}_2$  layer is in direct contact with the (p)a-Si:H locally or over the full area, and the low trap density state provided by the  $\text{SnO}_2$  layer is not sufficient for holes tunnelling from the (p)a-Si:H layer and electrons recombining in  $\text{SnO}_2$  thereby exhibiting the “S-shape” *JV* characteristic as observed. In terms of the effect of the 1.5 nm native  $\text{SiO}_2$  layer (as observed in Fig. 1D and Fig. S2, ESI†), trap assisted tunnelling is still feasible as shown in the simulated band diagram in Fig. S13 (ESI†).

To investigate the effect of ITO deposition, and ITO/ $\text{SnO}_2$ /perovskite stack deposition with varying ITO thicknesses on Si-surface passivation, and recombination, we carried out (i) photoluminescence (PL) imaging, and (ii) carrier lifetime, dark saturation current density and implied  $V_{OC}$  measurements on silicon half cells (front metallization not completed) before ITO deposition, after ITO deposition and after ITO/ $\text{SnO}_2$ /perovskite deposition.

Comparison of Si half cells before and after ITO deposition shows negligible changes in the photoluminescence intensity (Fig. S14, ESI†), carrier lifetime (Fig. S15A, ESI†) and implied  $V_{OC}$  (Fig. S15B and Table S4, ESI†) indicating that cells did not experience “sputter damage” from ITO deposition.

There were also negligible changes in the photoluminescence intensity (Fig. S14, ESI†), carrier lifetime (values remain in the same order of magnitude) (Fig. S15A, ESI†) and implied  $V_{OC}$  (Fig. S15B and Table S4, ESI†) of the Si half cells after the deposition of the ITO/ $\text{SnO}_2$ /perovskite, as  $\text{SnO}_2$  deposition required UV-ozone treatment prior to the ITO deposition. This indicates that the interfacing fabrication processes did not have a detrimental effect on the surface passivation quality of the Si half cells.

The effect of varying the interlayer ITO thickness on Si front-surface passivation was also negligible, with insignificant changes in photoluminescence intensities (Fig. S14, ESI†), carrier lifetimes (Fig. S15A, ESI†), implied  $V_{OC}$  (Fig. S15B and Table S4, ESI†) and dark saturation current density  $J_0$  (values remain in the same order of magnitude) (Table S5, ESI†). All of these indicate no change in the dominant recombination mechanism,<sup>38</sup> consistent with the shape of the injection independent lifetime curves remaining similar for Si cells with different ITO thicknesses (Fig. S16, ESI†).

It was found that our ITO is more conductive (Table S2, ESI†) than those typically used in SHJ solar cells.<sup>39</sup> For example, the resistivity of our ITO at 100 nm is  $4.28 \times 10^{-4} \Omega \text{ cm}$  ( $R_{SH}=42.8 \Omega$  per square) while for ITO normally used in a SHJ solar cell, the resistivity is around  $1.61 \times 10^{-3} \Omega \text{ cm}$  for a 70 nm thick layer

**Table 1** Characteristics of Si-perovskite  $1 \text{ cm}^2$  tandem test cells with varying ITO interlayer thicknesses prior to the application of anti-reflection layers

ITO (nm)	PCE (%)	$V_{OC}$ (mV)	$J_{SC}$ ( $\text{mA cm}^{-2}$ )	FF (%)	$R_S$ ( $\text{m}\Omega \text{ cm}^2$ )	$R_{SH}$ ( $\Omega \text{ cm}^2$ )	$R_{S, \text{Suns-Voc}}$ ( $\Omega \text{ cm}^2$ )	Pff (%)
0	15.5 ± 2.5	1758 ± 45	16.2 ± 0.4	52.0 ± 6.0	60 ± 10	6251 ± 1841	5.33	84.3
1.0	16.7 ± 1.7	1787 ± 33	16.1 ± 0.2	56.2 ± 5.2	35 ± 5	5820 ± 2630	5.59	83.8
1.7	22.1 ± 2.4	1775 ± 37	16.2 ± 0.6	77.6 ± 3.0	7 ± 3	3962 ± 937	0.82	84.4
2.4	20.8 ± 1.3	1753 ± 42	16.1 ± 0.3	73.4 ± 4.6	9 ± 4	3749 ± 686	0.89	84.0
3.5	16.8 ± 2.1	1663 ± 39	16.1 ± 0.2	61.2 ± 7.8	21 ± 10	1840 ± 750	3.26	70.4
5.0	13.9 ± 2.4	1603 ± 94	16.2 ± 0.3	53.5 ± 11.2	25 ± 8	1127 ± 411	8.61	65.8



Fig. 3 Simulated energy band structure of the tandem device under thermal equilibrium. Simulated band diagram of the full tandem device (A and B) with and (C and D) without ITO interface layers at equilibrium. (B) Zoomed-in view of (A) highlighting the ITO/(p)a-Si:H recombination stack. (D) Zoomed-in view of (C) highlighting the SnO<sub>2</sub>/(p)a-Si:H recombination stack. Solid and hollow red dots denote electrons and holes, respectively.

( $R_{SH} = 230 \Omega$  per square).<sup>39</sup> This can result in the formation a strong Schottky contact between the ITO and <p>a-Si:H layers leading to a depletion of charge carriers in the <p>a-Si:H and therefore a reduction in the built-in potential.<sup>40</sup> This needs to be compensated by the increase in the thickness of the <p>a-Si:H junction to increase the built-in potential back to desired levels.<sup>41,42</sup> This was experimentally confirmed by comparing representative silicon bottom cells with a 13 nm thick or 26 nm thick <p>a-Si:H layer as shown in Fig. S17 (ESI<sup>†</sup>). As expected, the cell with a 26 nm thick <p>a-Si:H produced higher  $V_{OC}$  and FF but lower  $J_{SC}$  due to higher parasitic short-wavelength absorption in the 26 nm thick <p>a-Si:H junction compared to the 13 nm thick <p>a-Si:H counterpart (Fig. S17C, ESI<sup>†</sup>). Tandem cell efficiency also improved when the 26 nm thick <p>a-Si:H layer junction was used (with an ITO interlayer fixed at 1.7 nm) as shown in Fig. S18 (ESI<sup>†</sup>) due to improvements in  $V_{OC}$  and FF. The thicker <p>a-Si:H layer junction no longer has an effect on tandem  $J_{SC}$  as short wavelength light is absorbed by the perovskite top cell while long wavelength light is absorbed deep in the bulk of the Si bottom cell. We then investigated the effect of varying the ITO interface layer thickness (0 nm, 1.0 nm, 1.7 nm, 2.4 nm, 3.5 nm and 5.0 nm) on the performance of tandem cells with a 26 nm <p>a-Si:H junction. The result shows

a similar trend to the cells with a thinner junction (Fig. S19, ESI<sup>†</sup>). That is, the optimum ITO interlayer thickness remained at 1.7 nm for the best power conversion efficiency.

After the application of the anti-reflection layer, we were able to demonstrate a 1 cm<sup>2</sup> tandem device (Fig. 4A and B) with a  $J_{SC}$  of 18.1 mA cm<sup>-2</sup>, a  $V_{OC}$  of 1.82 V, and a high FF of 82.4% under reverse scan and a steady state PCE of 27.0% with negligible hysteresis (Table 2). The PCE achieved is the highest for a n-i-p based perovskite/Si tandem solar cell demonstrating the effectiveness of this ultra-thin ITO interlayer design.

To demonstrate the scalability of our ultra-thin interface using ultra-thin ITO layers to large area, 11.8 cm<sup>2</sup> and 65.1 cm<sup>2</sup> cells were also demonstrated achieving efficiencies of 24.2% and 21.1%, respectively. The  $J-V$  and EQE curves of the champion cells are shown in Fig. 4. Their parameters are listed in Table 2. The 65.1 cm<sup>2</sup> tandem device produced a better infrared response due to the use of rear-textured cells while for the 1.0 and 11.8 cm<sup>2</sup> tandem devices, rear polished silicon cells were used purely due to the cell availabilities at the time of demonstrations. A video showing an encapsulated 4-inch size tandem cell at work powering a “mini-wind-turbine” can be found in the ESI<sup>†</sup> Performance drop in larger tandems was mainly caused by a decrease in FF. Reasons for higher hysteresis observed in larger



Fig. 4 Performance of champion tandem devices of different sizes. (A, C and E) light  $J$ - $V$  and (B, D and F) external quantum efficiency (EQE) of the perovskite-Si tandem devices of (A and B)  $1 \text{ cm}^2$ , (C and D)  $11.8 \text{ cm}^2$ , and (E and F)  $65.1 \text{ cm}^2$ . Insets show the photographs of the fabricated tandem devices and their steady-state current densities and steady-state PCEs at fixed voltages near their maximum power points.

tandem cells will need to be elucidated in future work. In terms of further  $J_{\text{SC}}$  improvements, simulated EQE spectra in Fig. S20A (ESI<sup>†</sup>) show that this can be done firstly by matching the current between the top and bottom cells increasing  $J_{\text{SC}}$  to around  $18.7 \text{ mA cm}^{-2}$ . After that major losses limiting  $J_{\text{SC}}$  shown in the pie chart in Fig. S20B (ESI<sup>†</sup>) need to be mitigated. They include front reflected and escaped losses as being the dominant (due to Fresnel reflection losses from refractive index mismatch between the top PDMS layer and the ITO/MoO<sub>3</sub>/Spiro-OMeTAD stack) loss followed by parasitic absorptions in the sun-facing Spiro-OMeTAD/MoO<sub>3</sub> stack. Loss reduction can be achieved by

replacing the stack with less absorbing materials<sup>43,44</sup> or switching to p-i-n configuration as shown in Fig. S20C and S20D (ESI<sup>†</sup>) thereby increasing  $J_{\text{SC}}$  towards  $20 \text{ mA cm}^{-2}$  for achieving a PCE of over 30%. In addition, efficiencies of large area cells can be further improved as metallisation and the fabrication of the perovskite layer continue to be optimised over large area to improve current and FF outputs.

The stability of an encapsulated tandem cell (Fig. S21, ESI<sup>†</sup>) was also tested under 1-sun continuous illumination with maximum power point (MPP) tracking. Encapsulation and testing details can be found in the ESI<sup>†</sup>. The results (Fig. 5)

**Table 2** Parameters of champion perovskite-Si tandem solar cells of various sizes with 26 nm thick (p)a-Si:H, 1.7 nm ITO interlayer, and after the application of anti-reflection layers

Area (cm <sup>2</sup> )	Scan	V <sub>OC</sub> (mV)	J <sub>SC</sub> (mA cm <sup>-2</sup> )	FF (%)	PCE (%)
1.0	Reverse	1819	18.1	82.4	27.2
	Forward	1804	18.1	80.6	26.4
11.8	Reverse	1771	18.0	76.0	24.2
	Forward	1770	17.9	74.5	23.7
65.1	Reverse	1777	17.6	67.5	21.1
	Forward	1772	17.7	62.3	19.5

show that the tandem cell can maintain 98% of its initial performance after 600 hours demonstrating excellent light stability of the device structure. Preliminary 60 °C testing in N<sub>2</sub> for 210 hours also shows reasonable thermal stability of the un-encapsulated tandem cell as shown in Fig. S22 (ESI†).

In conclusion, we have demonstrated the use of ultra-thin interface design for tandem solar cells with improved performance due to the reduction in parasitic resistance. As we are able to demonstrate perovskite-Si tandem devices with finer thickness resolution (*e.g.*, 0.5 nm), it may be possible in future work to observe non-linear behaviour which would likely to occur<sup>45</sup> with ultra-thin ITO interface layers when their thickness is varied at finer scale. In this work, a high fill factor (82.4% under reverse scan) was achieved using an interfacing ITO layer as thin as 1.7 nm. This value exceeds those reported thus far for perovskite-Si tandem devices. Power conversion efficiencies of 27.2%, 24.2% and 21.1% were achieved on 1.0 cm<sup>2</sup>, 11.8 cm<sup>2</sup> on 65.1 cm<sup>2</sup>, respectively, under reverse scan. The cell structure developed also

shows respectable stability with negligible efficiency loss after 600 hours of continuous 1 sun illumination at ambient temperature. This work has provided an elegant approach for optimizing interfacing for tandem solar cells. This is successfully demonstrated in this work with the integration of perovskite and Si solar cells for achieving large area monolithic tandem devices.

## Author contributions

J. Z. and A. H.-B. conceived the project. W. D., A. L. and K. D. fabricated the Si bottom cells. J. Z. integrated perovskites onto Si bottom cells and completed the monolithic tandem cell fabrication. J. Z. conducted device characterization. W. D. and K. C. F. assisted with silicon bottom cell analysis. Z. C. Z., Y. G., F. M. and D. R. M. carried out the device simulation. H. Y., S. T., J. B. and C. X. conducted film characterization. L. G. and M. J. assisted with tandem stability testing. J. Q., L. Y. and J. M. C. conducted the STEM and EELS measurements and the associated analysis. H. Y. and A. U. assisted with the MoO<sub>3</sub> layer deposition. C. Y., Y. Z., and S. B. assisted with the Suns-V<sub>OC</sub> and lifetime measurements for silicon bottom cells. A. H.-B., and D. R. M. supervised the research. J. Z. and A. H.-B. wrote the manuscript, and all authors were involved in the discussion of the results.

## Data and materials availability

All data needed to evaluate the conclusions in the paper are present in the paper or the ESI.† All the data are available from the corresponding authors upon reasonable request.



**Fig. 5** Stability testing of encapsulated 1 cm<sup>2</sup> perovskite-Si tandem cells under continuous illumination (100 mW cm<sup>-2</sup>) at relative humidity of 60 ± 20% at room temperature. Maximum power point tracking of (A) power output, (B) voltage (V<sub>MPP</sub>) at the maximum power point and (C) current to the starting value at the maximum power.

## Conflicts of interest

The authors declare that they have no competing interests.

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