



Fig. 2 Infrared absorption spectra of aniline phase-II at 550 K and pressures ranging from 31 to 33 GPa. (A) 750–1800 cm^{-1} spectral range and (B) 2750–3800 cm^{-1} spectral range. After reaching 33 GPa, a significant decrease in the intensity of aniline spectral features is observed, indicating the occurrence of a chemical reaction. (C) A kinetic curve representing the time evolution of aniline consumption at 33 GPa and 550 K. The full red line corresponds to the fit performed using eqn (2).

of-plane bending mode.²⁴ The absorption pattern was fitted using a Voigt profile and the total absorbance, the area of the band, was taken as a measurement of the amount of residual aniline during the evolution of the reaction. The data were analyzed with the Avrami model.^{25–27} A fitting equation derived from ref. 29, can be written as

$$A_t = A_{\text{inf}} + (A_0 - A_{\text{inf}})\exp[-k(t - t_0)^n] \quad (2)$$

where A_0 , A_t , and A_{inf} are the integrated absorptions of aniline at the beginning of the reaction (t_0), after a time t and at the equilibrium, respectively; n is a parameter that accounts for the growth dimensionality for a given nucleation law and k is the rate constant. The kinetic curve was nicely reproduced using the rate constant k and the n parameter reported in Fig. 2C. The n parameter can provide insight about the microscopic evolution of the reaction once it is related to the growth geometry and nucleation rate. An n value smaller than 1 unambiguously indicates unidimensional growth. In our case, the n parameter obtained was equal to 0.28, clearly indicating a product propagating with 1D growth. Similar n values were already reported in the pressure induced unidimensional polymerization of acetylene²⁸ and ethylene.²⁹

Once no further spectral modifications could be appreciated, the sample was first brought back to ambient temperature and then the pressure was slowly released. According to Fig. 2C, a certain amount of unreacted aniline, about 25%, was still present suggesting that lattice defects can prevent the 1D reaction propagation. Upon releasing the pressure, the intensity of all of the aniline bands decreased indicating that the reaction proceeds during the downstroke.

The rather homogeneous pale yellow/brownish material synthesized at high P and T and recovered at ambient conditions is shown in Fig. 3a. The infrared absorption spectrum of this material (Fig. 3b) resembles that of the product recovered



Fig. 3 (a) Optical images of the loaded aniline and the recovered pale yellow/brownish material after decompressing and opening the cell. (b) Infrared absorption spectra for selected pressures recorded along the decompression cycle at 298 K and infrared spectrum of the recovered material, lower trace.



from the high pressure reactivity of benzene thus suggesting a saturated material containing N–H bonds.²¹

Recently, a 1D, highly ordered, saturated nanomaterial with a local diamond-like structure was recovered after a controlled compression–decompression cycle of benzene up to 20 GPa.^{6,30} The product, characterized by a wealth of techniques like bright-field transmission electron microscopy and synchrotron X-ray diffraction, differed substantially from the amorphous material reported in previous reports,^{10,21} consisting of a tubular or thread-like structure, which was also supported by first principle calculations.^{6,30}

In view of the striking similarities with the benzene reactivity and the intriguing results of the Avrami model, suggesting a material characterized by 1D growth, the product was morphologically characterized using multiple techniques. The recovered material was mechanically removed from the gasket with a needle and placed directly onto the surface of a standard transmission electron microscope (TEM) copper grid. The collected multiple high-resolution TEM images at two different magnifications (Fig. 4a) exhibit parallel striations, suggesting the formation of threads or tubes. The line profile measured along the white line in the low magnification BF-TEM image (Fig. 4a – left) presented in Fig. 4b shows that the striations are ~ 5.5 Å apart. The high magnification BF-TEM image (Fig. 4a – right) evidenced the presence of long-range 1D parallel striations spaced at 4.0–5.1 Å and sizing tens of nanometers long. These values can be compared to the 1-D nanothreads obtained from benzene which were characterized by a distance between packed threads of ~ 6.4 Å.³⁰ The regions where the linear threads bend, as visible in the image with higher magnification (Fig. 4a – right), are likely related to crystal boundaries or dislocations of the starting aniline crystals. It should also be remarked that the nanothreads develop along a specific aniline crystal direction, namely the *a*-axis (see following discussion), which should therefore lie in the image plane to make the threads observable. The latter issue can account for the regions where apparently no threads are present.

In order to support the spectroscopic and morphological characterizations, quantum chemical calculations were performed using DFT to model a residue of a 1-D polyaniline-like nanothread structure consisting of 4-fused aniline molecules in which the sp^2 carbons of the rings were converted into sp^3 carbons by forming covalent bonds that develop between the rings. Views of the optimized geometry of a segment that accounts for the aniline-derived nanothread with the relative dimensions are presented in Fig. 4c. According to the DFT optimized geometry, the diameter dimensions expected for an aniline-derived nanothread are in the order comprised between 5.0 and 5.4 Å, which is in striking agreement with the parallel striations observed in the TEM images.

According to Thess *et al.*³¹ a 2D triangular lattice is characterized by a group of peaks in the low-Q region: a strong peak around 0.44 \AA^{-1} followed by four weaker peaks up to 1.8 \AA^{-1} . Angle dispersive X-ray diffraction measurement on the recovered material (Fig. 5a) notably agrees with the mentioned characterization, suggesting the formation of a triangular structure with an $a = 13.3 \text{ \AA}$, in the middle range of the lattice

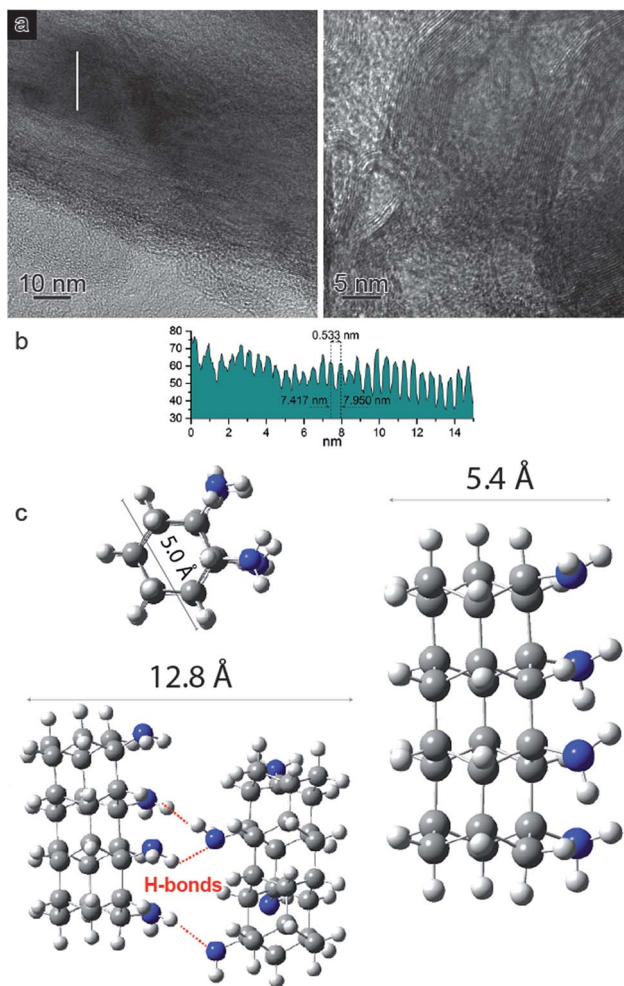


Fig. 4 (a) BF-TEM images at two different magnifications (see inset scale bars) of the recovered material presenting striations spaced at ~ 4 to 5.1 Å and extending in the longitudinal dimension tens of nanometers. (b) A line profile measured along the white line in the low magnification BF-TEM image (a – left). (c) Views of the optimized DFT geometry structure of the one-dimensional aniline-derived nanothreads with relative dimensions.

values reported for benzene-derived nanothreads (~ 6.4 Å)⁶ and fullerene single-wall carbon nanotubes (17 \AA).³¹ The lattice constant is two and half times the value of the nanothread diameter and almost double that of benzene derived nanothreads, which can suggest that the molecular orientation and the hydrogen bond network in the crystal strongly influence the reaction for aniline and the product characteristics. Crystal aniline presents a peculiar H-bond arrangement connecting the NH_2 groups of the nearest neighbor molecules and developing along the *a*-axis.^{15,32} The presence of such strong interactions in the crystal prevents the participation of the NH_2 groups in the reaction and favors the remarkably anisotropic compression along the direction more suitable for inter-ring interaction. The natural conclusion is that these constraints selectively drive the inter-ring formation of C–C bonds, with the consequent C hybridization change from sp^2 to sp^3 , along the *a*-axis. Therefore, double nanothreads form along this direction interacting



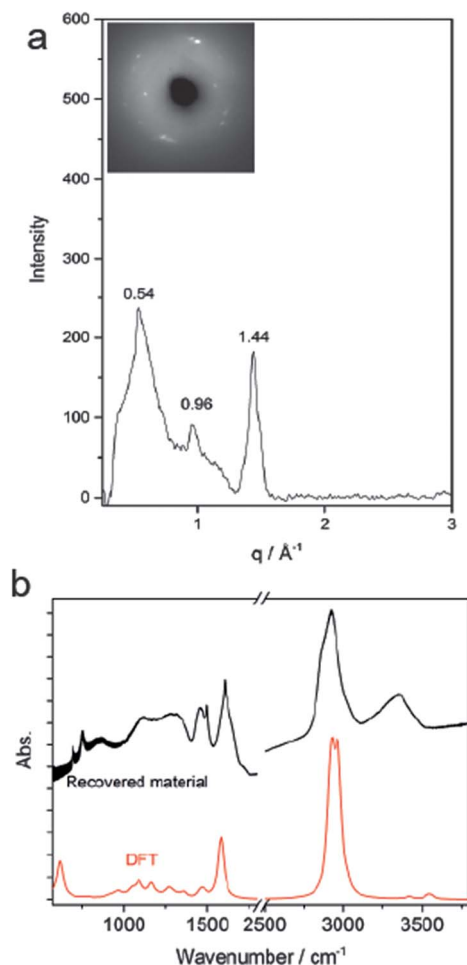


Fig. 5 (a) Bragg peaks in the total X-ray scattering structure. The inset shows the original 2D diffraction image of the recovered material. (b) Upper black trace: IR absorption spectra measured at ambient conditions of the recovered sample removed from the DAC. Lower red trace: calculated IR spectra for the 1D aniline-derived nanothreads. Band shapes were modelled using Gaussian functions with a bandwidth of 20 cm^{-1} .

through H-bonds and having a size of $\sim 12.8\text{ \AA}$ (Fig. 4c), resulting in an expanded 2D triangular lattice when compared to benzene. This is in excellent agreement with the lattice parameter ($a = 13.3\text{ \AA}$) derived from the angle dispersive X-ray diffraction pattern.

Finally, the structure of the one-dimensional polyaniline nanowire with a local diamondoid-like geometry was optimized and its vibrational spectrum was calculated by quantum chemical calculations using density functional theory (DFT). Two views of the optimized nanowire geometry and one view of the double nanowire interacting through the H-bond region are presented in Fig. 4c. DFT calculations strongly support the formation of one-dimensional aniline-derived nanowires with the structural parameters and the calculated infrared spectrum being in great accordance with the experimental data. Both IR spectra, calculated and measured, are dominated by the C–H stretching modes of sp^3 hybridized carbon atoms at 2900 cm^{-1} and by the bending N–H modes at

1575 cm^{-1} (Fig. 5b) thus providing solid evidence for the formation of a one-dimensional aniline-derived nanowire.

The properties of different functionalized diamond nanowires have been recently computed by DFT calculations.³³ These materials correspond to the possible products of high-pressure transformations of both functionalized benzene and heteroaromatic rings. Interestingly, the functionalized diamond nanowires maintain the mechanical properties of the pristine material but offer, depending on the functional group and its spatial distribution, the possibility of tuning the band gap. According to these predictions the NH_2 -enriched carbon polyaniline-like nanowire is expected to present a band gap in the order of 3.5 eV , an essentially insulating material due to the intrinsic carbon sp^3 character, an ideal strength of $\sim 14.8\text{ nN}$, a Young's modulus of 163 nN and a fracture strain (ϵ_{max}) of ~ 0.16 .³³ Moreover, the synergic effect between these remarkable mechanical properties with the versatility of the NH_2 groups decorating the exterior of these nanowires representing potential active sites for doping and as linkers for molecules with biological interest and inorganic nanostructures, must be taken into account.

Methods

Aniline ($\text{C}_6\text{H}_5\text{NH}_2$, Merck) was distilled under reduced pressure prior to use and was loaded into a MDAC (membrane diamond anvil cell) equipped with IIA type diamonds and a rhenium gasket where a hole with an initial diameter of $150\text{ }\mu\text{m}$ was drilled and used as a sample chamber. In order to reduce the strong IR absorption of the sample, the optical path was reduced by pressing KBr into the sample chamber producing a pellet whose surface was successively scratched. Afterwards, liquid aniline and a ruby chip were added above the KBr pellet resulting in a sample thickness ranging from 10 to $20\text{ }\mu\text{m}$. High-temperature experiments were performed using the resistively heated MDAC. The temperature was measured with an accuracy of $\pm 0.1\text{ K}$ by a K-type thermocouple placed close to the diamonds. FT-IR absorption measurements were recorded with an instrumental resolution of 1 cm^{-1} using a Bruker-IFS 120 HR spectrometer modified for high-pressure measurements.¹⁶ The ruby fluorescence was excited using a few milliwatts of a 532 nm laser line from a Nd:YAG laser source.

Angle-dispersive X-ray diffraction (ADXRD) experiments were performed at the ESRF high-pressure beamline ID27 using monochromatic X-ray radiation of wavelength $\lambda = 0.3738\text{ \AA}$, a MARCCD 165 detector positioned at 146 mm from the sample, as calibrated with a CeO_2 standard. The focal spot (fwhm) of the beam was $\sim 3\text{ }\mu\text{m}$. The 2D diffraction patterns were integrated using DIOPTAS; manual background subtraction was done in Fityk.

Bright-field imaging (BF-TEM) was performed using an FEI Titan Themis 60-300 transmission electron microscope operated at an 80 kV accelerating voltage. The sample was removed from the gasket with a needle directly onto the surface of a standard TEM copper grid.

Quantum chemistry calculations were performed using the Gaussian03 package¹⁷ to obtain optimized structures and



- 20 S. Fanetti, M. Citroni and R. Bini, *J. Chem. Phys.*, 2011, **134**, 204504.
- 21 L. Ciabini, M. Santoro, R. Bini and V. Schettino, *J. Chem. Phys.*, 2002, **116**, 2928–2935.
- 22 W. Li, D. Duan, X. Huang, X. Jin, X. Yang, S. Li, S. Jiang, Y. Huang, F. Li, Q. Cui, Q. Zhou, B. Liu and T. Cui, *J. Phys. Chem. C*, 2014, **118**, 12420–12427.
- 23 P. W. Bridgman, *Phys. Rev.*, 1914, **3**, 153–203.
- 24 H. M. Badawi, W. Förner and S. A. Ali, *Spectrochim. Acta, Part A*, 2013, **112**, 388–396.
- 25 M. Avrami, *J. Chem. Phys.*, 1939, **7**, 1103–1112.
- 26 M. Avrami, *J. Chem. Phys.*, 1940, **8**, 212–224.
- 27 M. Avrami, *J. Chem. Phys.*, 1941, **9**, 177–184.
- 28 M. Ceppatelli, M. Santoro, R. Bini and V. Schettino, *J. Chem. Phys.*, 2000, **113**, 5991–6000.
- 29 D. Chelazzi, M. Ceppatelli, M. Santoro, R. Bini and V. Schettino, *J. Phys. Chem. B*, 2005, **109**, 21658–21663.
- 30 B. Chen, R. Hoffmann, N. W. Ashcroft, J. Badding, E. Xu and V. Crespi, *J. Am. Chem. Soc.*, 2015, **137**, 14373–14386.
- 31 A. Thess, R. Lee, P. Nikolaev, H. Dai, P. Petit, J. Robert, C. Xu, Y. H. Lee, S. G. Kim, A. G. Rinzler, D. T. Colbert, G. E. Scuseria, D. Tomanek, J. E. Fischer and R. E. Smalley, *Science*, 1996, **273**, 483–487.
- 32 N. P. Funnell, A. Dawson, W. G. Marshall and S. Parsons, *CrystEngComm*, 2013, **15**, 1047–1060.
- 33 J. F. R. V. Silveira and A. R. Muniz, *Phys. Chem. Chem. Phys.*, 2017, **19**, 7132–7137.

