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Geometry, mechanics and actuation of intrinsically curved folds†

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We combine theory and experiments to explore the kinematics and actuation of intrinsically curved folds (ICFs) in otherwise developable shells. Unlike origami folds, ICFs are not bending isometries of flat sheets, but arise via non-isometric processes (growth/moulding) or by joining sheets along curved boundaries. Experimentally, we implement both, first making joined ICFs from paper, then fabricating flat liquid crystal elastomer (LCE) sheets that morph into ICFs upon heating/swelling via programmed metric changes. Theoretically, an ICF's intrinsic geometry is defined by the geodesic curvatures on either side, κ_g . Given these, and a target 3D fold-line, one can construct the entire surface isometrically, and compute the bending energy. This construction shows ICFs are bending mechanisms, with a continuous family of isometries trading fold angle against fold-line curvature. In ICFs with symmetric κ_g , straightening the fold-line culminates in a fully-folded flat state that is deployable but weak, while asymmetric ICFs ultimately lock with a mechanically strong finite-angle. When unloaded, freely-hinged ICFs simply adopt the (thickness t independent) isometry that minimizes the bend energy. In contrast, in LCE ICFs a competition between flank and ridge selects a ridge curvature that, unusually, scales as $t^{-1/7}$. Finally, we demonstrate how multiple ICFs can be combined in one LCE sheet, to create a versatile intrinsically curved gripper that lifts a heavy weight.

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Curved folds lend strength, form and function to surfaces throughout biology and engineering (Fig. 1), and have been extensively studied in the beautiful and useful art of origami.^{6,11–14} However, not all curved folds are equal. The defining mechanics of thin sheets is that they are essentially inextensible, as isometric bend has a much lower elastic cost than in-plane stretch. Origami artists bend flat paper, making their folds extrinsic (isometric to the plane) but there is no such restriction on folds formed by stitching flat sheets,^{15,16} or by non-isometric processes such as differential growth^{17–19} or moulding. Such processes can thus create intrinsically curved folds (ICFs), which can only be flattened by stretch, giving richer geometry and stronger mechanics. Here, we present the basic kinematic rules for describing and classifying ICFs, which provide insight into their (lack of) rigidity, and also into their utility as mechanisms and deployable structures.

An additional motivation for ICFs stems from the topical field of “metric mechanics”,²⁰ which studies flat actuating sheets that morph into intrinsically curved surfaces. Such morphing requires a programmed spatial pattern of actuation,

reminiscent of differential growth during morphogenesis, and has been implemented with swelling-gels,²¹ phase-changing



Fig. 1 Curved folds in biology, architecture, and engineering. Top: Leaf of *Salix babylonica* ‘Annularis’ (S–),¹ giant water lily (A+),² carambola fruit (S+, S–).³ Middle: LCE ribbon (S–), Cupola, Sedgwick Museum (S+),⁴ origami tower (D. Huffman, S–).^{5–7} extrinsic. Bottom: Top hat (A–),⁸ tent (A+),⁹ umbrella (S+).¹⁰ Labels (SA/±) show sign/symmetry of the geodesic curvatures.

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liquid crystal elastomers (LCEs),^{22–24} dielectric elastomers²⁵ and pneumatic baromorphs.²⁶ The generation of intrinsic (Gauss) curvature gives dramatically strong actuation: for example, LCE disks that morph into conical shells can lift 1000× their weight.²⁷ Thus inspired, we also demonstrate how to program an LCE sheet to morph into an ICF,^{28,29} leading to flat ribbons that macroscopically “bend” into arcs but *via* strong Gaussian actuation.

1 Geometry of intrinsically curved folds

The elementary way to fabricate an ICF is to stitch together two flat sheets of material along curved boundaries. The seam then forms an ICF through an otherwise developable surface (Fig. 2). This approach is routinely used by engineers, architects and tailors to create intrinsically curved surfaces from flat material. Before stitching, the boundary of each flat piece ($i = 1, 2$) may be assigned a local outward normal $\hat{\mathbf{m}}_i$ and curvature vector $-\kappa_{g_i}\hat{\mathbf{m}}_i$, so that $\kappa_{g_i} > 0$ is an inward curvature vector and *vice versa*. After stitching, both boundaries must follow the same (arc-length parameterized) 3D space curve $\mathbf{r}(l)$ which defines the fold line. The resultant ICF geometry is characterized by the fold's curvature vector $\boldsymbol{\kappa} = \mathbf{r}''(l)$, and, in a cross-section perpendicular to the fold-line tangent, $\mathbf{r}'(l)$, the fold angles α_i between $\boldsymbol{\kappa}$ and each flank. As marked in Fig. 2, $\alpha > 0$ indicates a clockwise rotation from $\boldsymbol{\kappa}$ to flank (around the tangent), so the total fold angle is $\alpha_1 - \alpha_2$.

The geodesic curvature κ_g of a curve on a surface is computed as the projection of its 3D curvature vector into the tangent plane. However, a foundational result in differential geometry dictates that κ_g is an intrinsic quantity that is invariant under isometric deformations.³⁰ Accordingly, in the ICF, the geodesic curvatures of the original flat boundaries, κ_{g_i} , must match the projection of $\boldsymbol{\kappa}$ into the respective flank:

$$\kappa_{g_1} = |\boldsymbol{\kappa}| \cos \alpha_1, \quad \kappa_{g_2} = |\boldsymbol{\kappa}| \cos \alpha_2. \quad (1)$$

For any proposed fold line with $|\boldsymbol{\kappa}| \geq \max(|\kappa_{g_1}|, |\kappa_{g_2}|)$, one may apply these kinematic relations to compute $\alpha_i(l)$. Furthermore, since the flanks are developable and hence ruled, their form is then determined, even far from the fold. Precisely, the angle

$\beta_i(l)$ between ruling and fold (Fig. 2), may be computed (ref. 12, ESI,† Section S1) in terms of the fold torsion $\tau(l)$:

$$\cot \beta_i = \frac{\alpha'_i(l) - \tau(l)}{|\boldsymbol{\kappa}(l)| \sin \alpha_i(l)}. \quad (2)$$

Another celebrated intrinsic property of surfaces is Gaussian curvature, K , computed as the product of the two principal curvatures. An ICF's Gaussian curvature defies direct computation due to the sharp apex. However, the Gauss–Bonnet theorem allows us to compute the total curvature $\Omega = \int K dA$ in any patch of surface from the geodesic curvature of its boundary. A simple application to ICFs gives the distribution of total curvature along the fold as²⁹

$$\frac{d\Omega}{dl} = \kappa_{g_1} + \kappa_{g_2} = |\boldsymbol{\kappa}|(\cos \alpha_1 + \cos \alpha_2), \quad (3)$$

so any fold with $\kappa_{g_1} \neq -\kappa_{g_2}$ is intrinsic with singular K .

2 Kinematics and classification of curved folds

Eqn (1) reveals that the kinematics of an ICF are governed by its two geodesic curvatures. To explore what deformation are permitted, we focus on homogeneous ICFs, made by joining pieces with constant κ_{g_i} (annular sectors) along the arc of a circle, leading to uniform fold angles. Such ICFs can also be fabricated straightforwardly by joining annular arcs of craft paper (ESI,† Section S6), enabling observations of kinematics and mechanics. Five ICF categories emerge, based on combinations of κ_{g_i} , as illustrated in Fig. 3 and Movies M1–M5 (ESI†).

Regular surfaces are often classified by the sign of their Gaussian curvature, with $K = 0$ (flat), $K > 0$ (cap-like) and $K < 0$ (saddle-like) giving different geometry and mechanics. Accordingly, we first consider a $K = 0$ fold (origami case in Fig. 3) which requires $\kappa_{g_1} = -\kappa_{g_2} \equiv \kappa_g$. The two initial pieces are thus

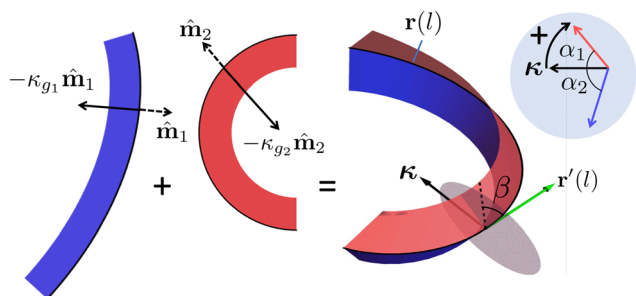


Fig. 2 Construction of a general ICF.

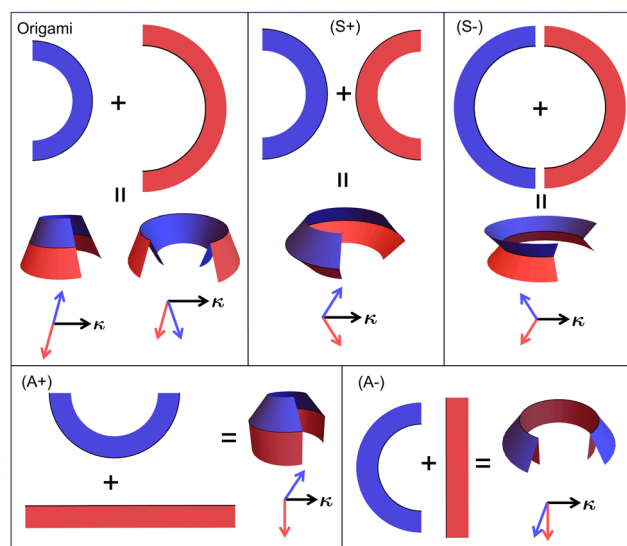


Fig. 3 Curved-fold origami and the classification of ICFs.



complimentary, and can fit together whilst flat, confirming the fold is extrinsic. The kinematic constraints (eqn (1)) give the fundamental rules of origami folds,^{11,12} $\alpha_1 = \cos^{-1}(\kappa_g/|\kappa|)$, and $\alpha_2 = \pi - \alpha_1$. The least curved state, $|\kappa| = \kappa_g$, is the completely flat state ($\alpha_1 = 0$ and $\alpha_2 = \pi$), while, during folding, increasing curvature creates an increasingly sharp fold, with the fold-line bi-normal bisecting the flanks (Movie M1, ESI†).

Alternatively, an elementary $K > 0$ fold can be made by taking $\kappa_{g1} = \kappa_{g2} = \kappa_g > 0$, a symmetric positive ICF (S+). Eqn (1) now requires $\alpha_1 = -\alpha_2 = \cos^{-1}(\kappa_g/|\kappa|)$, so κ itself bisects the flanks, ensuring an equal projection into each (Fig. 3 (S+), Movie M2, ESI† and Fig. 1 cupola/umbrella). The least curved state, $|\kappa| = \kappa_g$, is a flat-folded closed-book configuration, and increasing $|\kappa|$ requires the book to open, tending towards an unfolded state, $\alpha_1 = -\alpha_2 = \pi/2$, as $|\kappa|$ diverges. The fold line and apex curvature have the same sense, as expected for $K > 0$ surfaces, and these isometries can be interpreted as trading fold-line curvature and folding angle at constant K . A corresponding symmetric negative fold (S−) is made as $\kappa_{g1} = \kappa_{g2} \equiv -\kappa_g < 0$, giving a fold with almost identical behavior (Fig. 3 (S−), Movie M3, ESI† and leaf in Fig. 1) except the book-like state is inverted $\alpha_1 = -\alpha_2 = \pi$ and, as the curvature increases the flanks approach the unfolded $\alpha_1 = -\alpha_2 = \pi/2$ from above.

However, ICFs are not fully characterized by their Gaussian curvature, as eqn (1) applies to each flank individually. Thus folds with asymmetric curvatures (A±) behave differently. We focus on two prototypical examples made by joining a straight strip, $\kappa_{g1} = 0$ with an annulus of each sign $\kappa_{g2} = \pm\kappa_g$ (Fig. 3 A+ and A−, Movies M4, M5, ESI†). Either way, the folding condition on the straight strip requires $\alpha_1 = \pi/2$ so κ has zero projection. The least curved states still have $|\kappa| = \kappa_g$, giving $\alpha_2 = 0, -\pi$ respectively, so the A+ fold resembles a capped cylinder (Fig. 1 lilly pad), and A− a flanged pipe (Fig. 1 top hat). Thus, asymmetric ICFs have finite fold angles even in their least curved state. As previously, increasing $|\kappa|$ causes unfolding, with the highly curved but completely unfolded state $\alpha_1 = \pi/2$, $\alpha_2 = -\pi/2$ reached as $|\kappa|$ diverges.

The essential kinematic feature of ICFs is that fold angle dictates curvature and vice-versa, giving a continuous family of isometries in which the two change in tandem (Movies M1–M5,

ESI†). ICF's are thus simple mechanisms, and a small actuator controlling fold angle can be used to manipulate the fold's curvature. Furthermore, if the fold-angle is fixed then any deviation of the curvature from its isometric value is strongly resisted. A simple illustration of this mechanism principle is that a symmetric positive fold fashioned from paper can be used as a simple but effective grabber (Fig. 4A and Movie M6, ESI†).

For symmetric folds, unbending of the fold-line (reducing $|\kappa|$) culminates in a flat-folded state. Symmetric folds are thus attractive deployable structures, which can be constructed whilst flat, then unfurled into a 3D Gauss-curved surface.

The different ICF categories also have different mechanical responses. If a load seeks to unbend an asymmetric ICF, it will deform isometrically until it locks rigidly in its least curved state—a 3D shape with finite fold angle—with further unbending requiring stretch. In contrast, unbending a symmetric fold culminates in the flat-folded state that, mechanically, is a single floppy sheet that can easily buckle to accommodate further unbending. This pattern is confirmed by simple tensile tests (Fig. 4B). Interestingly, both A+ and A− folds do eventually buckle out of plane into inhomogeneous ICF geometries, but A+ are considerably stronger than A−. The key difference is that further unbending requires tensile stretches in the annular flank of A+, but compression in A−, which promote buckling. A+ folds are thus particularly suitable for applications requiring rigidity or strong actuation.

3 Flank bending energy

An ICF formed by joining inextensible flat sheets is limited to the isometric conformations discussed above. However, these configurations are not elastically equivalent, as they have different bends in the flanks. Quantifying this bend energy allows one to assess which isometry will be observed whilst unloaded, and the ICF's stiffness as it deforms. When a flank is bent from the initial flat state to form an ICF, it stores elastic energy $E = \int \frac{1}{2} D k^2 dA$, D being flexural rigidity, k the single finite curvature of a developable surface, and the integral is

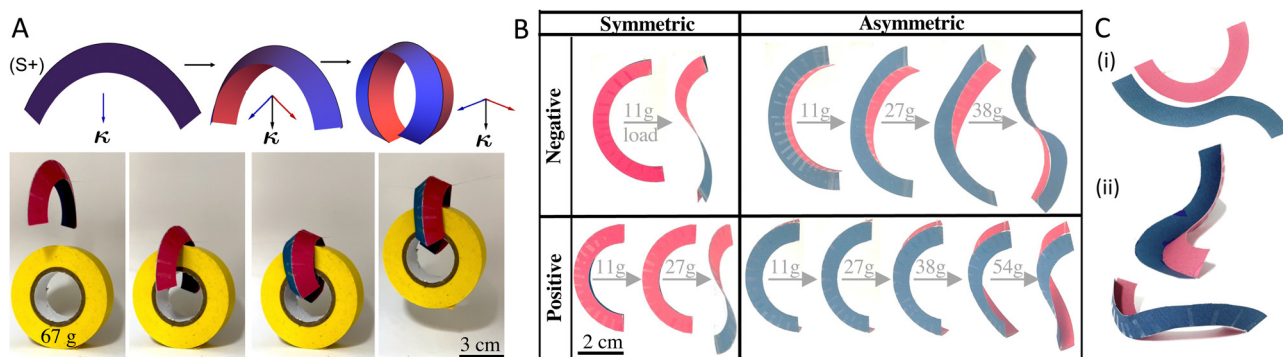


Fig. 4 (A) Bending mechanism of the symmetric positive (S+) fold trading curvature vs fold angle, and a corresponding paper grabber. (B) Experimental tensile strength measurements for the four types of folds. (C) (i) Paper strips in 2D with in-homogeneous and asymmetric κ_g ; (ii) Resulting ICF has a curved ridge with non-zero torsion.



over flank area. In the special case of a homogeneous ICF, the whole ICF is a surface of revolution with conical flanks connected by a circular fold. In this case, the bending energy of an individual flank (per-unit length of fold) may be precisely evaluated as

$$\rho_{fi} = \frac{D}{2}(|\kappa|^2 - \kappa_{gi}^2) \frac{\log(1 - \kappa_{gi} w_i)}{-\kappa_{gi}} \approx \frac{D w_i}{2}(|\kappa|^2 - \kappa_{gi}^2), \quad (4)$$

where w_i is the width of the flank, and the second form is accurate in the narrow flank limit. As expected, these expressions are zero if $|\kappa| = \kappa_{gi}$, when the flank lies flat, and penalize curvature in excess of this. If the fold of the ICF is treated as a free-hinge, then the resting configuration will minimize the bending energy of the two flanks, giving $|\kappa| = \max(|\kappa_{g1}|, |\kappa_{g2}|)$. In symmetric folds, this corresponds to the fully closed state, while in asymmetric ICFs it gives the locked limiting state. These states are indeed observed as the unloaded equilibria of our paper ICFs.

In general, eqn (2) allows one to reconstruct the flanks of any ICF, and then, following Wunderlich,^{31–34} compute the associated bend energy as a 1D fold-line integral containing $|\kappa|$ and τ (ESI,[†] Section S2). In the narrow limit, this procedure gives

$$\rho_{fi} = \frac{D}{2} w_i \frac{(|\kappa|^2 - \kappa_{gi}^2 + (\tau - \alpha'_i)^2)^2}{|\kappa|^2 - \kappa_{gi}^2}, \quad (5)$$

where α_i follows from $|\kappa|$ via eqn (1). The unloaded form of the ICF then follows by minimizing the energy of both flanks over τ and $|\kappa|$. Interestingly minimizing over τ gives a simple local condition, which reveals that homogeneous ICFs ($\alpha'_1 = \alpha'_2 = 0$) and symmetric ICFs ($w_1 = w_2$, $\kappa_{g1} = \kappa_{g2} \Rightarrow \alpha'_1 = -\alpha'_2$) will form torsion-free plane curves. However inhomogeneous asymmetric ICFs generically do have torsion in their minimizing configuration, generating non-planar fold lines (Fig. 4C).

4 Intrinsically curved folds via metric mechanics

ICFs in biological tissues are not formed by stitching of flat sheets, but by patterns of differential growth that directly alter a single sheet's metric.¹⁷ The emerging field of metric mechanics^{21,35} provides an enticing engineering analogue, by programming patterns of shape-change into flat sheets of soft actuating material. ICFs formed *via* growth or metric-mechanics follow the same kinematics as their joined counterparts, but with the added feature of actuating from flat, and different mechanics stemming from their non-hinged fold. To investigate these differences, we fabricate flat LCE ribbons that morph into ICFs upon stimulation, due to a spatially programmed molecular alignment direction $\mathbf{n}(x,y) \equiv (\cos \psi(x,y), \sin \psi(x,y))$. The actuation mechanism is that heating or swelling the LCE disrupts this alignment (mirroring the conventional nematic–isotropic phase transition) and causes a large uniaxial shape change, with markedly different stretching factors $\lambda_{\parallel}$ along $\mathbf{n}$, and $\lambda_{\perp}$ in the orthogonal direction, $\mathbf{n}^{\perp}$ (Fig. 5A and B).

In metric terms, this means an infinitesimal vector $d\mathbf{l} = (dx, dy)$ in the flat sheet changes length from $d\mathbf{l}^2 = d\mathbf{l} \cdot d\mathbf{l}$ to $d\mathbf{l}_A^2 = d\mathbf{l} \cdot \bar{\mathbf{a}}(x,y) \cdot d\mathbf{l}$, where the new metric has the form

$$\bar{\mathbf{a}}(x,y) = \lambda_{\parallel}^2 \mathbf{n}(x,y) \otimes \mathbf{n}(x,y) + \lambda_{\perp}^2 \mathbf{n}^{\perp}(x,y) \otimes \mathbf{n}^{\perp}(x,y). \quad (6)$$

Previous work has considered flat LCEs programmed with pairs of circle^{28,36} or spiral²⁹ director patterns, and shown that an inhomogeneous ICF of finite extent appears along the interface between patterns upon actuation. Typically, these ICFs resemble a mountain pass between the conical tips generated by the circle/spiral centers. Here, we instead seek a transitionally invariant pattern to produce an infinitely extendable homogeneous ICF. We thus consider a flat LCE ribbon that extends infinitely in the x direction, and has a transitionally invariant director profile $\psi(y)$ (Fig. 5B). Following our treatment of joined

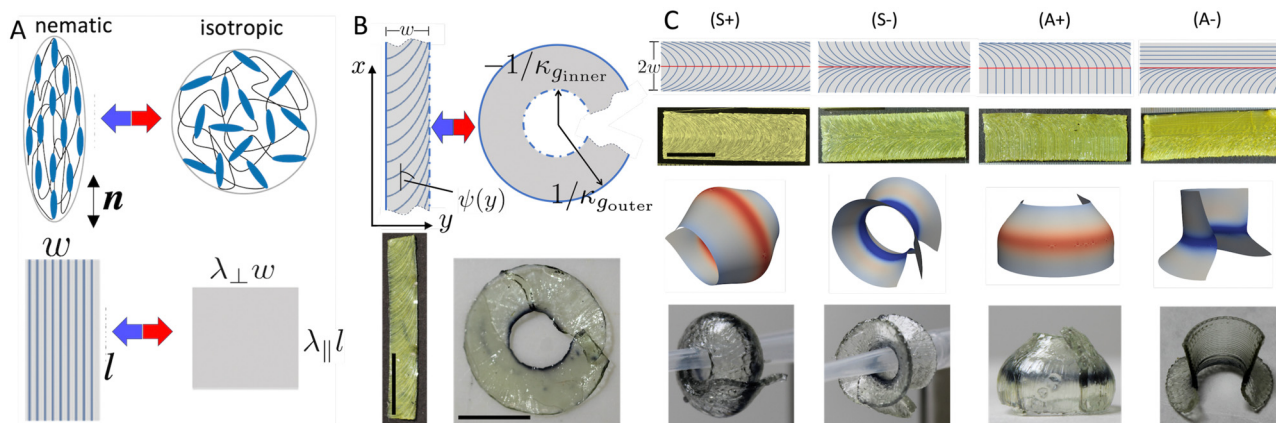


Fig. 5 (A) The nematic–isotropic transition in an LCE, which can be induced by heating or swelling with an isotropic solvent. The monodomain nematic director is $\mathbf{n}$. (B) Top: A planar director pattern in an LCE ribbon that morphs it into an annular arc on actuation. Bottom: 3D printed LCE following the arc-pattern before and after actuation by swelling in toluene (unactuated thickness ≈ 0.2 mm, 2 printed layers). (C) The four types of ICF can be encoded in an LCE by combining pairs of arc patterns (S+/-), or the arc pattern with a monodomain (A+/-). From top to bottom: Theoretical director patterns, unactuated experimental strips, and actuated ICFs in simulation and experiment. Experiments and simulations have matched (unactuated) thickness ≈ 0.4 mm (4 printed layers), $w = 5$ mm, and actuation factors $\lambda_{\parallel} \approx 0.9$, $\lambda_{\perp} \approx 2.4$. All scale bars are 10 mm.

ICFs, we first seek a profile that morphs the ribbon into an annular sector. Since such a sector remains Gaussian flat during actuation, we may apply the Theorema Egregium and set $K = 0$ to find the profile $\psi_c(y) = \frac{1}{2} \arccos(2y/w - 1)$,³⁷ where w is the width of the ribbon, and the profile varies from along $\hat{x}$ to $\hat{y}$ over the ribbon's width. A direct computation (ref. 38, ESI,† Section S4) gives the geodesic curvatures of the strip's boundaries as

$$\kappa_{g_{\text{inner}}} = \frac{\lambda_{\parallel}^2 - \lambda_{\perp}^2}{2w\lambda_{\parallel}^2\lambda_{\perp}}, \quad \kappa_{g_{\text{outer}}} = \frac{-\lambda_{\parallel}^2 + \lambda_{\perp}^2}{2w\lambda_{\parallel}^2\lambda_{\perp}},$$

confirming they follow circular arcs, and hence that the strip becomes annular. Here $\kappa_{g_{\text{outer}}} > 0$ and $\kappa_{g_{\text{inner}}} < 0$ follow the sign convention in Fig. 2. As sketched in Fig. 5C, we may then form positive and negative symmetric ICFs by combining pairs of patterns in a single sheet, with the join becoming the ICF. Similarly, asymmetric ICFs can be created by combining a single pattern with a ribbon of uniform director.

To verify these designs, we fabricated LCE ribbons *via* extrusion-based 3D printing³⁹ (materials and methods), using the extrusion direction to encode the spatial alignment pattern, and each printed layer adding $\approx 100 \mu\text{m}$ of thickness. After printing, actuation was tested on mono-domain ribbons. Swelling in toluene produced actuation factors of $\lambda_{\parallel} \approx 0.9$ and $\lambda_{\perp} \approx 2.4$ (Fig. S10, ESI†), while thermal actuation yields $\lambda_{\parallel} \approx 0.5$ and $\lambda_{\perp} \approx 1.3$ by 130°C (Fig. S9, ESI†). Swelling of a ribbon printed with a single copy of the pattern $\psi_c(y)$ indeed produces an annular arc, and ribbons with the four pair-wise combinations of patterns indeed produce the four categories of ICF (Fig. 5B and C). The actuated shapes of the ICFs were also computed numerically using the bespoke active-shell C++ code Morphoshell,³⁸ producing excellent agreement with the experiments.

5 Relaxed shape of metric-mechanic ICFs

Interestingly, the LCE ICFs do not adopt the simple bend-minimizing forms discussed above: for example the symmetric

ICFs are far from fully closed. The key difference is that now the fold is not a freely jointed hinge, so the elastic sheet resists the singular bending deformations required to create a sharp apex. Indeed, one may readily observe in experiments and numerics that the central fold is not sharp, but rather blunted over some lengthscale f , that is short compared to flank-width and radius, but not compared to thickness. Such blunting requires a competition between stretch and bend, with small strains (deviations from isometry) occurring to relieve the singular curvature of a sharp ridge. To capture this competition, we consider a homogeneous ICF with fixed curvature $|\kappa|$, as created in our experiments. A perfect isometry of the ICF would form a surface of revolution consisting of two conical flanks connected by a sharp ridge at radius $R_0 = 1/|\kappa|$, which we may describe in cylindrical coordinates as $\bar{R}(s) = R_0 - |s|\cos\alpha_i$, with s being the arc-length transverse to the fold, $s = 0$ being the apex, and α_i switching value between the flanks in accord with eqn (1). Similarly, we describe the blunted form of the ICF by the smooth curve $R(s) = \bar{R}(s) + \Delta R(s)$, and also define the $\theta(s)$ as the angle between the local tangent and the radial direction (see Fig. 6(A)). During blunting, the dominant bend-stretch competition is between s curvature $\theta'(s)$ (which would diverge at a sharp fold) and hoop strain, $\varepsilon = \Delta R/\bar{R}$, leading to the simplified shell energy density

$$W = \frac{1}{2}Y\left(\frac{\Delta R}{\bar{R}}\right)^2 + \frac{1}{2}D\theta'(s)^2, \quad (7)$$

where $Y = 3\mu t$, $D = \frac{1}{3}\mu t^3$ are stretching and bending moduli respectively, μ being the (incompressible) LCE's shear modulus, and t the actuated thickness. Assuming that the length-scale of blunting is short compared to R_0 , and also that the isometry is nearly cylindrical so that $\theta'(s) = \Delta R''(s)$, we may write the energy of the ridge as

$$E \approx L \int \frac{1}{2}Y\left(\frac{\Delta R}{R_0}\right)^2 + \frac{1}{2}D\Delta R''^2 ds, \quad (8)$$

where L is the length of the ICF. Minimizing variationally with respect to ΔR requires $\Delta R^{(4)}(s) + Y\Delta R/DR_0^2$, which admits four

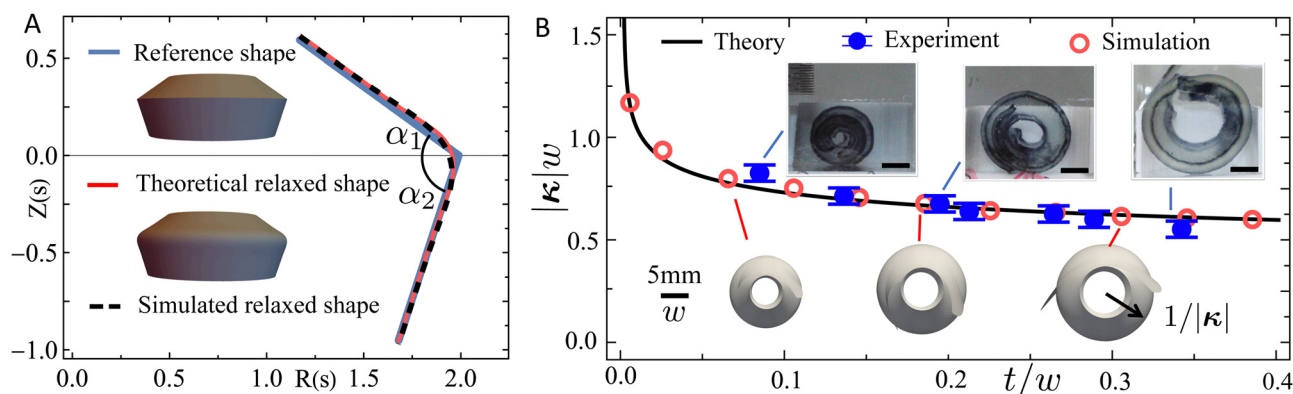


Fig. 6 (A) Reference (isometric) shape of an ICF with $\alpha_1 = 0.2\pi$ and $\alpha_2 = 0.4\pi$, and its theoretical (eqn (9)) and simulated relaxation. (B) Equilibrium ICF curvature against thickness for an (S+) fold: theory (eqn (11)), experiments, and simulations. All simulation and experiment have equal planar dimensions (reference half-width $w = 50 \text{ mm}$ and length 3 cm) and span a range of actuated thicknesses t .

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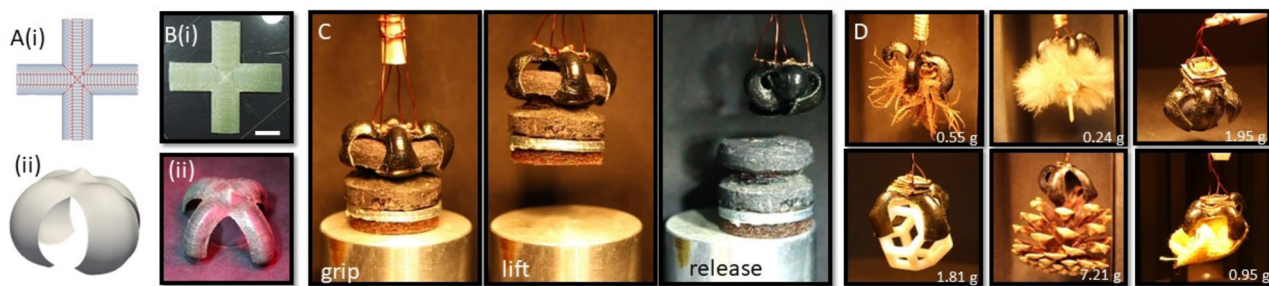


Fig. 7 The design of an LCE Gaussian gripper. A (i) Reference director pattern, encoding several ICFs. (ii) Simulated actuated configuration makes a strong claw shape. (B) 4-layer 3D printed sample weighing 0.48 g (i) before and (ii) after actuation with an IR heat lamp. (C) Controlled grip, lift and release of a weighted cork. Release occurs shortly after the stimulation is turned off. (D) The gripper can lift a wide range of object: clockwise, Turkish hazelnut husk, silverpuff, blueberry, fabric, pinecone and 3D printed dodecahedron frame. Scale bar on B (i) is 1 cm.

fabricate an LCE Gaussian-gripper, by combining multiple strong asymmetric positive folds in a cross-like configuration, (Fig. 7A and B). Upon convenient thermal actuation by a heat lamp, the cross morphs into a claw shape that is capable of gripping and lifting a simple load (Fig. 7C) up to 40× the gripper's own weight (Movie M7, ESI†). Although this weight multiple is naturally scale dependent, it considerably exceeds the performance of previous bend-driven soft grabbers of comparable dimensions.^{39,48} The gripper is also versatile, lifting a wide range of objects with different shapes, weights, textures and levels of compliance (Fig. 7D and Movie M8, ESI†).

Looking ahead, our results also motivate many directions for further exploration. Can one design an ICF that traces an arbitrary 3D space curve in its equilibrium state, or even an elephant's-trunk ICF that can morphs between many space-curves *via* angular actuators along its length? What is the effective rod-theory^{47,49} for ICF ribbons? How do ICF kinematics change when they pass through intrinsically curved ribbons?⁵⁰ What are the kinematics of surfaces containing multiple ICFs, such as our gripper? Overall, the rich geometry and mechanics of ICFs coupled with their straightforward manufacture, suggests these questions, any many others, will be a very fruitful area for further exploration.

7 Materials and methods

7.1 Materials

LC ink was synthesised by creating oligomers *via* aza-Michael addition from a reactive mesogen (RM82; CAS: 125248-71-7), amine linker (*n*-butylamine, *n*BA; CAS: 109-73-9) and photoinitiator (Ir2959; CAS: 106797-53-9). RM82 was purchased from Daken Chemical, China, and both *n*BA and Ir2959 from Merck Life Science, UK. All compounds were put to a glass vial (with molar ratio of RM82 to *n*BA: 1.1:1.0 and with 1 wt% of initiator), melted with a heat gun and vigorously mixed, first by vortex mixer and then on a hot plate with magnetic stirrer bar for *ca.* 30 min. A (pre-heated) metal syringe for printing was then filled with the ink and left in the oven for around 18 h for full oligomerisation (75–80 °C).

7.2 3D-printing of ICFs

The syringe with the ink was mounted in Hyrel's KR2-15 used with System 30 M 3D printer with a 2D array of UV LED (365 nm). All the patterns were printed on glass slides coated with 5% solution of poly(vinyl alcohol) in water, after careful calibration of the printing parameters for 0.30 mm MK8 nozzle (printed line width: 0.20 mm, first layer height: 0.125 mm, layer height: 0.1 mm, printing speed: 260 mm min⁻¹, extrusion rate: 5.2 μL min⁻¹, priming: 40 000–50 000 pulses with rate 10 000 pulses per s, where 1297 pulses correspond to 1 μL; temperature of printing: 80 °C), with UV LEDs working on 30–50% of max power. After printing all samples were additionally cured for 1 h in a UV curing station with two UV LEDs (LuxiGen LZ1-10UV0R, 365 nm) irradiating a sample placed on a distance of 65 mm from both top and bottom diode. Printed LCE sheets were then detached from their glass substrates with a blade.

7.3 Actuation

Initial thermal strain data was collected on monodomain (linear) samples, using a hot plate (Fig. S9, ESI†). Swelling strain data was obtained by placing monodomain samples in a glass container filled with toluene for around 3 h (Fig. S10, ESI†). ICF samples were actuated by swelling in the same way, and then imaged using a Canon EOS 1200D camera, with either EFS 18–55 mm lens or with macro 100 mm lens. ICFs in Fig. 5C were imaged shortly after removal from toluene. The precisely characterized ICFs in Fig. 6B were imaged whilst still in toluene, with the camera directly facing the axis of the fold's surface of revolution for accurate extraction of the ICF radius. Thickness, before and after actuation, was measured using digital calipers (Hilka 0–150 mm). The LCE grippers were actuated thermally with a heat lamp (Panasonic, 300 W) in a custom box coated internally with 0.4 mm thick black paper. The LCE sample was prepared by 3D printing, then painted with a black marker to aid absorption, and the lamp was directed at an angle of 30° from the vertical. The same camera was used to record experiments of grasping and lifting different objects with a weight range 0.24–19.24 g. Temperatures up to 150 °C were recorded during heat lamp irradiation, using a



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