

MINIREVIEW

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Synthesis and application of polypyrrole nanofibers: a review

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State-of-the-art polypyrrole nanofiber-based nanoarchitectonics can be generally fabricated by electrospinning, interfacial polymerization and reactive template methods. Even though analogous nanofiber morphologies and nanofibrous network architectures can be obtained by these methods, the structural details and structural complexities may alter significantly as different synthesis methods are applied. For the electrospinning technique, on one hand, nanofibers can be directly obtained by spinning polypyrrole-containing dope solutions; on the other, the electrospun nanofiber mats can be used as templates to direct the nanofiber formation; a two-step fabrication process, including the electrospinning of polymer nanofiber mats and deposition of polypyrrole on the polymer nanofibers' surface, is generally employed. By tuning the electrospinning parameters, the composition, diameter, morphology, and alignment of the as-obtained electrospun nanofiber mat can be effectively controlled, which may allow the fabrication of polypyrrole nanofibers with sophisticated nanostructures and nanoarchitectures. Interfacial polymerization is capable of generating polypyrrole nanofibers without templates. It is speculated that the protonation and re-orientation of polypyrrole at the oil–water interface may decoil the polymer chains and transform them into more extended conformations, while the charged polymer chains more easily diffuse into the water phase and form a stable dispersion. Different from electrospinning, the reactive templates may drive the formation of polypyrrole nanofibers through either redox or protonation mechanisms.

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Nanofibers with different curvatures, compositions, and architectures can be obtained by using different types of reactive template in a simple, fast, environment-friendly and one-step manner. A wide range of applications have been demonstrated by the polypyrrole nanofiber-based nanoarchitectonics, including cell culture, tissue engineering, neural stimulation, energy storage, and organic electronics.

1. Introduction

Nanofibers of the intrinsically conductive polymer polypyrrole have demonstrated their applied functions spanning a wide range of engineering fields, *e.g.*, supercapacitors, sensors, electromagnetic shielding, tissue engineering, neural stimulation, and neural regeneration.^{1–6} The superior functional performance of polypyrrole nanofibers can be ascribed to their high specific surface area, interpenetrating network structure, high electrochemical activity, tunable band gap, and biocompatibility. Different techniques have been applied to synthesize polypyrrole nanofibers, including electrospinning, interfacial polymerization, soft templating, and seeding.^{7–10} In order to obtain polypyrrole nanofiber products with high specific surface area, appropriate electronic structure, and good mechanical properties, the length, diameter, flexibility, morphology, and conductivity of the nanofibers should be controlled carefully and effectively. Moreover, the as-obtained polypyrrole nanofibers can be assembled into 1D, 2D, and 3D microstructures, *e.g.*, microfibers, sheets and membranes, in a continuous manner, making them attractive for the large-scale production of nanoarchitectonics with hierarchical nano-micro-geometries.¹¹ Different from conventional nanotechnology, nanoarchitectonics feature the harmonization of multiple, multiscale, and complex processes and interactions in the molecular, nano, and micro-scale to create life-like morphologies and functions.¹² Recently, new polypyrrole nanoarchitectonics with different micro-nano-structures and interfacial compositions, such as polypyrrole/metal halide/DNA hybrids,¹³ polypyrrole hollow foam loaded with high-density MOF nanowires,¹⁴ and polypyrrole:polystyrene sulfonate/cellulose nano-paper,¹⁵ have emerged, showing promising applications in biosensors and wearable supercapacitors. Generally speaking, the fabrication of polypyrrole nanofibers may involve the synergy of free-radical polymerization, hydrophobic interaction, formation of a nano-micro-interface by nanofiber self-assembly *etc.* Harmonization of the synthesis processes may create nanofiber architectures with fractal geometries, where self-similar fibrous structures can be observed both microscopically and macroscopically. However, the existing techniques are diverse in the mechanisms, materials, and methods to produce polypyrrole nanofibers, so a review of and comparison between the existing techniques are necessary for understanding their characteristics, pros and cons, and differences, in order to rationally design the structural and nano-architectural features of polypyrrole nanofibers for engineering applications.

2. Electrospinning

Electrospinning has been widely used to fabricate electrospun nanofiber mats of different polymeric materials, providing the polymer can be properly dissolved in a solvent to make a dope

solution.¹⁶ During the electrospinning process, the dope solution is loaded into a reservoir equipped with a blunt-needle-type outlet. To start electrospinning, the dope solution is ejected through the outlet at a constant flow rate by applying pressure and a voltage of up to 50 kV is applied to the needle. As a result of the applied voltage, charges will accumulate on the surface of the liquid meniscus at the needle tip, ejecting a continuous liquid jet from the meniscus, forming a hydrodynamic structure called a Taylor cone.¹⁷ The liquid jet will move towards the collector, which is grounded or negatively charged, and placed several centimeters away from the needle tip. During the movement, the liquid jet will split, stretch, and solidify, forming nanofibers that randomly deposit on the surface of the collector; a schematic illustration of the electrospinning process of nanofibers is shown in Scheme 1. To obtain nanofiber mats with desired morphologies, structures, pore size distributions, sizes, and thicknesses, the electrospinning parameters, including the viscosity, surface tension, applied voltage, flow rate, needle size, needle/collector distance, collector dimensions, temperature and relative humidity, should be controlled carefully. Electrospinning might be the most versatile technique for fabricating nanofibers with different compositions, morphologies, structures and functions. It has been used extensively to fabricate different types of polypyrrole nanofibers, and the polypyrrole nanofiber membranes obtained from electrospinning have been used in a wide range of applications, *e.g.*, cell culture, tissue engineering, cancer therapy, biosensors, and supercapacitors.^{18–22}

However, pure polypyrrole nanofibers cannot be obtained directly when the electrospinning techniques are used. Additional coating steps to form a polypyrrole layer on the surface of the electrospun nanofibers are required when polymer solutions not containing polypyrrole are used as the dope solution for electrospinning. On the other hand, pure polypyrrole is difficult to be electrospun into nanofibers due to its low solubility and dispersability. Additives such as co-solvents,



Scheme 1 Illustration of the electrospinning process.

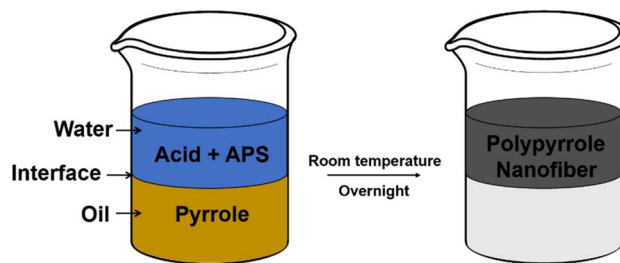


surfactants, and polymer carriers are needed to prepare polypyrrole solutions that are suitable for electrospinning,^{23,24} which may significantly alter the conductivity of the nanofibers. Many attempts have been made to circumvent the obstacle of polypyrrole solubility, for example, polypyrrole with improved solubility in organic solvents was synthesized by the addition of functional dopants, *i.e.*, [di(2-ethylhexyl)sulfosuccinic acid sodium salt], butylphenylthene sulfonic acid sodium salt, and dodecylbenzene sulfonic acid sodium salt;²⁵ upon further functionalization with chlorosulfonic acid, the water solubility of the as-obtained polypyrrole can reach 3 wt%, due to which it can be utilized for the electrospinning process in the aqueous phase.²⁶ E. Tavakkol *et al.* reported the fabrication of electrospun polypyrrole nanofibers by spinning the mixture solution of pyrrole and poly(vinylpyrrolidone) (PVP) into an ethanolic solution of FeCl₃.²⁷ The addition of PVP into the spinning dope solution can remarkably increase its viscosity and enhance the spinnability, but a second polymerization step was required to convert the pyrrole/PVP nanofibers into polypyrrole/PVP nanofibers, which may significantly affect the diameters and conductivities of the nanofibers.

Even though electrospinning has been proven to be one of the most versatile techniques to generate polymer nanofiber membranes and composites, it may require precise control over a wide range of processing parameters to achieve continuous and bead-free production of nanofibers, and the tuning of the parameters, *e.g.*, applied voltage, viscosity, temperature and relative humidity, is tedious and empirical, which may lower the efficiency of the fabrication process. High applied voltages and organic solvents are used during the electrospinning process, which may make it energy-consuming and environment-polluting. Nanofibers obtained from electrospinning may have a relatively wide range of size distribution and it is hard to obtain monodisperse nanofibers.

3. Interfacial polymerization

Polypyrrole nanofibers can be directly obtained from the free-radical polymerization of pyrrole through interfacial polycondensation. In a typical process, pyrrole is dissolved in an organic phase, *e.g.*, xylene, carbon disulfide, carbon tetrachloride, chloroform, methylene chloride, or toluene, while a strong acid (*e.g.*, 1 M hydrochloric acid) and a persulfate-type initiator [*e.g.*, ammonium persulfate (APS)] are mixed homogeneously to form the aqueous phase. To initiate the interfacial polymerization process, the water phase is carefully transferred to the top of the organic phase in a container, or *vice versa*, depending on the difference in density between these two phases. Due to the immiscibility of the two phases, an oil–water interface would form and the formation of the polypyrrole nanofibers may occur by the diffusion of the pyrrole monomer from the organic solvent to the aqueous phase *via* the oil–water interface. The interfacial polymerization can proceed at room temperature and the reaction occurs overnight to obtain polypyrrole nanofibers with sufficient degrees of polymerization and well-developed nanofibrous morphologies. Moreover, the as-obtained nanofibers can be assembled into a nanofibrous

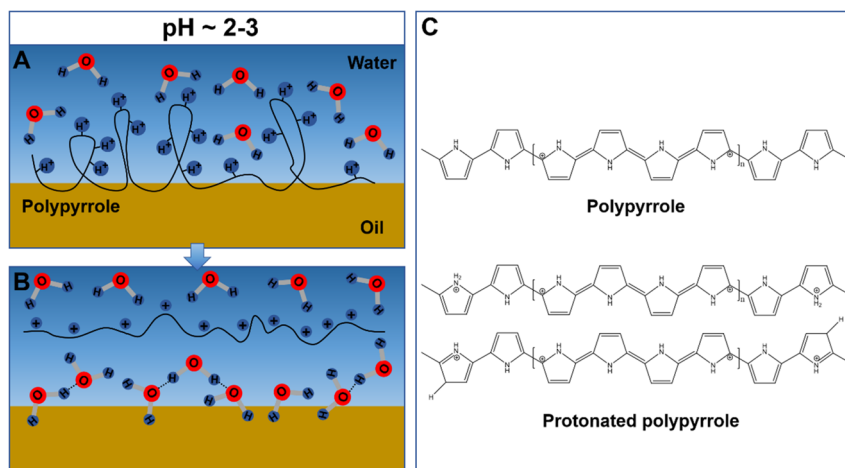


Scheme 2 Illustration of the synthesis of polypyrrole nanofibers by interfacial polymerization.

membrane by using a simple suction-filtration process. A schematic illustration of the interfacial polymerization process is shown in Scheme 2.

Compared to the fabrication of polypyrrole nanofibers by using electrospinning, where the electrospun nanofibers can be used as the hard template, no specific templates are required for the fabrication of polypyrrole nanofibers *via* interfacial polymerization. It is speculated that the formation of the nanofibrous morphology is closely related to the adsorption, orientation and desorption of the polypyrrole molecules at the oil–water interface and is highly pH-dependent. The pH-dependent adsorption/desorption process of polyelectrolytes has been investigated by D. K. Beaman *et al.*, who showed the change in pH can significantly alter the conformation of the polyelectrolytes, as a result of the increased charge density in the polymer backbone; the charge accumulation in the backbones would induce the intra/inter-molecular repulsion of the polymer, driving the swell and de-coil of the polymer chains, forming more extended and linear conformations; the charges in the polymer backbone can promote the ionization of the neighboring groups and reduce the free energy of desorption, and the entire polymer chain may move to the bulk water when a substantial portion of it is solvated and the free energy of desorption is sufficiently low.²⁸ The formation of the polypyrrole nanofibers through interfacial polymerization can also be explained by the pH-dependent adsorption/desorption of the polypyrrole chains at the oil–water interface. Briefly speaking, when the water phase containing acids and initiators is spread on top of the oil phase containing pyrrole, the pyrrole monomers adsorbed at the oil–water interface would be oxidized by the persulfate ions, forming pyrrole radicals and polymerizing through free radical polymerization. Simultaneously, the as-obtained polypyrrole chains would be protonated by the protons adsorbed at the oil–water interface. Q. Pei *et al.* reported that polypyrrole can be partially protonated in an aqueous solution with pH around 2–3,²⁹ which is consistent with the pH value of the aqueous phase used for the interfacial polymerization. As a result of the electrostatic repulsion, the protonated polypyrrole chains would adopt a more linear and extended conformation instead of the random coil conformation, which may promote the formation of nanofibers; the charges distributed along the protonated polypyrrole chains can reduce the free energy of desorption and promote the solvation of the as-formed nanofibers in the bulk water phase. Short and





Scheme 3 pH-dependent adsorption and desorption of polypyrrole at the oil–water interface. Polypyrrole is firstly protonated by the protons (H^+) at pH $\sim 2-3$ (A), and subsequently adopts an extended conformation as a result of the electrostatic interaction, and desorbs from the oil–water interface due to a reduction in free energy (B). (C) The chemical structures of the pristine polypyrrole and protonated polypyrrole.

curved nanofibers with small diameters (10–80 nm) can be formed by using interfacial polymerization.^{30–32} However, by harmonizing the electrochemical and interfacial polymerization techniques, polypyrrole nanofiber arrays with controllable nanofiber lengths can be obtained.³³ A schematic illustration of the pH-dependent adsorption/desorption process of polypyrrole at the oil–water interface is shown in Scheme 3.

The addition of surface-active reagents, *e.g.*, surfactants, during the interfacial polymerization may significantly alter the morphology of the polypyrrole nanofibers. Recently, Z. Hou *et al.* investigated the effect of the hydrophilic–lipophilic balance (HLB) value on the morphology of polypyrrole synthesized by interfacial polymerization.³⁴ The HLB value was tuned by mixing different proportions of Span80 and OP10 to make a mixture surfactant. The mixture surfactant and $FeCl_3$ were dissolved in water to make the aqueous phase, while pyrrole was

dissolved in chloroform to make the oil phase. As the HLB value of the mixture surfactant increased from 4.3 to 14.5, the morphology of the as-obtained polypyrrole changed from ruptured vesicle-like (Fig. 1A and B) to network-like (Fig. 1C and D) and to coral-like (Fig. 1E and F), respectively. The formation of the network-like structure at HLB = 10.4 indicated that the mixture surfactant in this case was more effective in de-coiling and stabilizing the polypyrrole chains at the interface. However, the formation of a primitive network-like structure instead of a well-defined nanofibrous interpenetrating network in the absence of a strong acid indicates that the protonation and extension of the polypyrrole chains under acidic conditions is vital in promoting the nanofiber formation.

The morphology of polypyrrole synthesized by interfacial polymerization is very sensitive to the synthesis conditions. H. Albaris *et al.* reported the use of interfacial polymerization to

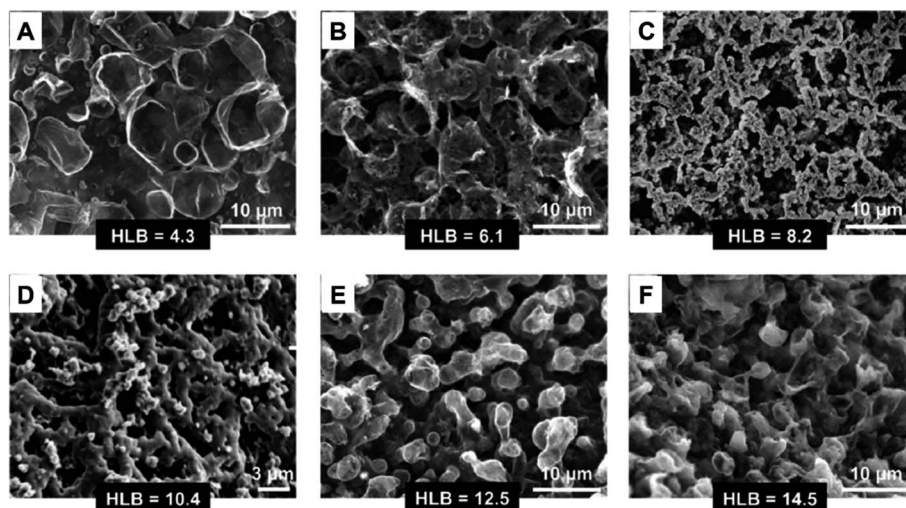


Fig. 1 Changes in the morphology of polypyrrole synthesized by interfacial polymerization with the addition of mixture surfactants with different HLB values. The morphologies of the as-synthesized polypyrrole alter from ruptured vesicle-like (A and B) to network-like (C and D), and eventually to coral-like (E and F) as the HLB value of the surfactant increased. Reproduced from ref. 34, with permission from Elsevier, 2018.



synthesize polypyrrole and polyaniline.³⁵ During the synthesis, pyrrole or aniline was dissolved in carbon tetrachloride along with HCl and cetyltrimethylammonium bromide (CTAB) to make the oil phase, while APS was dissolved in water to make the aqueous phase. However, only particulate aggregates were obtained after the polymerization, indicating the addition of HCl and CTAB in the oil phase was unfavorable for the nanofiber formation. This phenomenon can be mainly attributed to the polymerization of pyrrole catalyzed by diluted HCl,³⁶ which may generate polydisperse polypyrrole nanospheres as the nucleation centers to induce the growth of polypyrrole particles. Polypyrrole nanofibers can also be obtained by conducting the interfacial polymerization at the interfaces formed between the oil droplets and bulk water. J. Hazarika *et al.* reported the synthesis of polypyrrole nanofibers by dispersing the xylene droplets containing pyrrole into the water phase containing *p*-toluenesulfonic acid (*p*-TSA) and APS under constant stirring in ambient conditions.³⁷ Compared to the nanofibers synthesized by using a static oil–water interface, the dispersed oil–water interface created by the constant stirring may result in a lower monomer concentration, a higher proton concentration, and a higher initiator concentration at the local oil–water interface between the oil droplets and the bulk water phase, where an increased polymerization rate can be observed, and a faster mass transport from the oil–water interface to the bulk water can also be expected, and eventually the formation of short nanofibers with small diameters was observed.

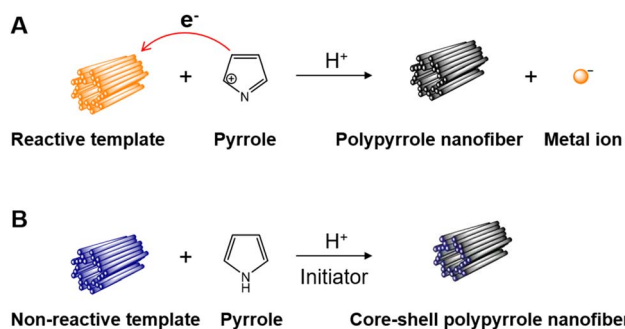
4. Solution-based template methods

Polypyrrole nanofibers can be readily synthesized in aqueous solutions by using templates with 1D nanofibrous morphologies. The solution-based template methods have been proven to be simple, fast, and high-efficiency routes to fabricate polypyrrole nanofibers with versatile compositions, morphologies, and practical functions.³⁸ Based on the reactivity of the 1D nanofibrous templates with respect to the polymerization reaction of pyrrole, the templates can be categorized as reactive or non-reactive. During the polymerization process, the reactive templates can not only act as the replica molds to direct the formation of nanofibers, but can also serve as the initiators to trigger the polymerization of pyrrole. On the other hand, the

non-reactive templates can only serve as the replica molds for forming the nanofibers. A schematic illustration of the solution-based template method is shown in Scheme 4.

To date, ionic complexes and metal oxides have been mainly used as the reactive templates, and are capable of producing polypyrrole nanofibers that perfectly reflect their primitive morphologies. In 2005, A. Wu *et al.* reported the synthesis of polypyrrole nanofibers by using an ionic complex composed of CTAB and APS as a reactive template.³⁹ The synthesis was conducted in an aqueous solution containing 1 M HCl. A dense polypyrrole-nanofiber network composed of long and curved nanofibers can be obtained by using this method. The diameters of the nanofibers were in the range between 20 and 40 nm, and the nanofibers were randomly oriented, showing large curvatures (Fig. 2A). Formation of the highly curved polypyrrole nanofibers can be possibly attributed to the preferential 2D stacking of the CTAB-APS ionic complexes in the aqueous solution with a low pH,⁴⁰ which may form loosely packed lamellar micelles as the templates for nanofiber growth (Fig. 2B and C). However, to maintain the structure of the lamellar micelles, excess APS (molar ratio of CTAB:APS = 1:3) is required. Recently, K. Shu *et al.* developed a new “tandem” strategy to fabricate flexible thin-film composites of exfoliated graphene and polypyrrole nanofibers.⁴¹ “Tandem” means the CTAB used to fabricate the exfoliated graphene is also used for the synthesis of polypyrrole nanofibers without removal. To obtain the composite of exfoliated graphene and polypyrrole nanofibers, pyrrole and concentrated HCl were added into the dispersion containing exfoliated graphene and CTAB, and subsequently APS was added to initiate the polymerization process. The polymerization was proceeded in an icy bath under magnetic stirring for 6 h, and subsequently flexible thin-film composites of exfoliated graphene and polypyrrole nanofibers can be obtained by simple suction filtration and washing processes. The as-obtained thin-film composite showed a thickness of around 10 μm , which was composed of stacking layers of polypyrrole nanofibers intercalated with graphene sheets. Compared to the thin film composed solely of exfoliated graphene or polypyrrole nanofibers, the composite film showed enhanced mechanical properties, specific capacitance, and rate performance, which were important for supercapacitors.

On the other hand, nanostructured metal oxides that are able to initiate the free radical polymerization of pyrrole can be used to fabricate nanofibrous polypyrrole with different morphologies, compositions, and architectures. It has been recently reported by D. P. Dubal *et al.* that polypyrrole nanotubes, nanofibers, and urchins can be obtained by using manganese dioxide (MnO_2) nanorods, nanofibers, and urchins as the reactive templates.⁴² The synthesis process for the nanostructured polypyrrole was simple and straightforward: the nanostructured MnO_2 was firstly dispersed in an aqueous solution of 1 M HCl, where pyrrole and the oxidant ($\text{K}_2\text{Cr}_2\text{O}_7$) were subsequently added and the reactants were properly mixed. The polymerization was conducted at room temperature and nanostructured polypyrrole with the same morphologies as the MnO_2 templates can be readily obtained (Fig. 3A and B). The MnO_2 templates were simultaneously removed as the



Scheme 4 Illustration for the synthesis process of polypyrrole nanofibers using (A) a reactive template and (B) non-reactive template.



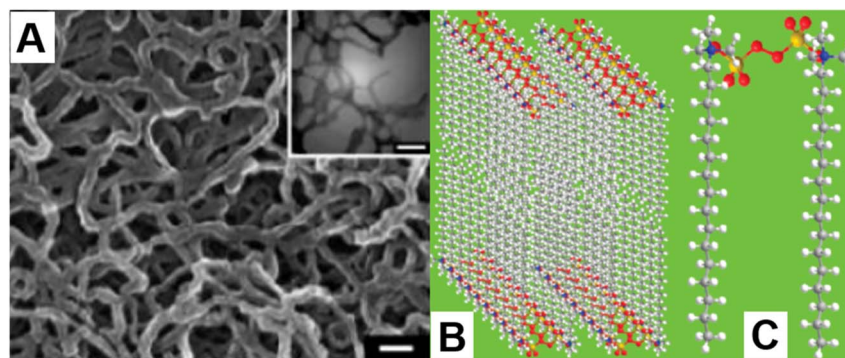
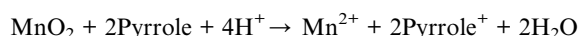


Fig. 2 Synthesis of polypyrrole nanofibers using the CTAB-APS ionic complex. (A) The morphology of the as-synthesized nanofibers. Reproduced from ref. 39, with permission from The American Chemical Society, 2005. (B) The 3D lamellar structure of the CTAB-APS complexes and (C) the 3D structure of a single CTAB-APS complex. Reproduced from ref. 40, with permission from The American Chemical Society, 2010.

polymerization process was completed, which can be attributed to the displacement reaction between MnO_2 and pyrrole. As MnO_2 has a higher oxidation potential ($E = +1.224 \text{ V}$)⁴³ than pyrrole ($E = +0.7 \text{ V}$),⁴⁴ it can be readily reduced by pyrrole, forming soluble Mn^{2+} cations and pyrrole radicals, which can be written as:



Excellent specific capacitance, cycling stability and rate performance were observed for the nanostructured polypyrrole synthesized using MnO_2 as the reactive template due to the increased electrochemically active surface area, fast ion/electron transport, and enhanced interfacial redox reactions. For the nanostructured polypyrrole with different

morphologies, the nanofibers showed the best electrochemical properties for energy storage applications, possibly due to their ability to assemble into interconnected network architectures. Other than nanofibers solely composed of polypyrrole, core-shell nanofibers composed of a metal oxide core and a polypyrrole shell can also be obtained by using the reactive template. J. Wang *et al.* reported the fabrication of V_2O_5 @polypyrrole core-shell nanowires by using V_2O_5 nanowires as the reactive template.⁴⁵ The V_2O_5 nanowires were firstly synthesized by a hydrothermal method and then dispersed in an aqueous solution of sodium dodecylbenzenesulfonate (NaDBS). The pH of the dispersion was adjusted to 1 using diluted H_2SO_4 and the temperature was maintained at $0-5^\circ\text{C}$ by using an icy bath. Subsequently the polymerization process was started by adding pyrrole, which was oxidized by the V_2O_5

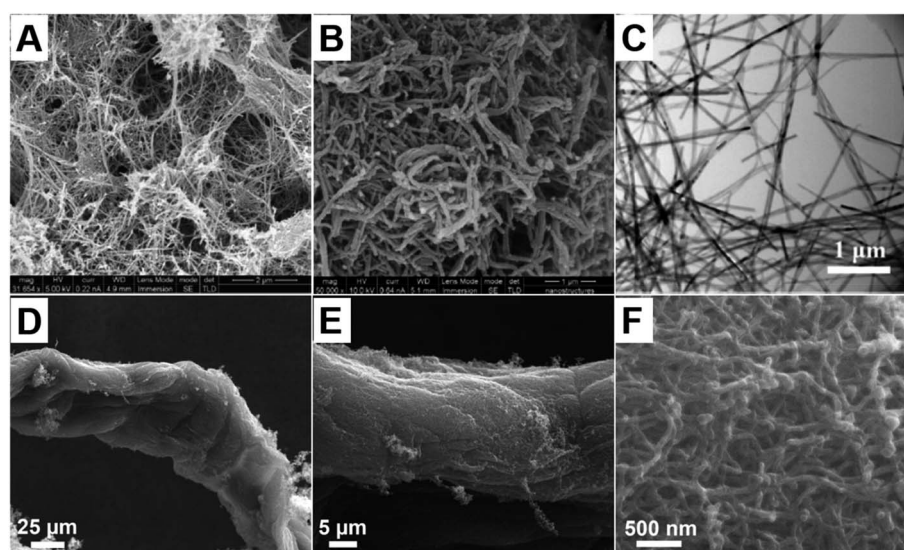


Fig. 3 Polypyrrole nanofibers synthesized by using nanostructured metal oxides as the reactive template. The morphologies of the MnO_2 nanowire templates and the as-synthesized polypyrrole nanofibers are shown in (A) and (B), respectively. Reproduced from ref. 42, with permission from Elsevier, 2016. (C) TEM image showing the core-shell structure of the V_2O_5 @polypyrrole nanowires. Reproduced from ref. 45, with permission from The American Chemical Society, 2018. (D) SEM image showing the morphology of the bulk polypyrrole fiber; microscale creases and folds can be observed on the fiber surface due to the applied stress during fiber formation. (E) The magnified morphology of the folding area, and (F) the morphology of the polypyrrole nanofibers that constitute the bulk fiber. Reproduced from ref. 46, with permission from The American Chemical Society, 2022.



nanowire templates and formed a polypyrrole layer on the surface of the nanowires. The TEM image of the V_2O_5 @-polypyrrole core-shell nanowires is shown in Fig. 3C, where the crystalline V_2O_5 cores and the amorphous polypyrrole shells can be clearly distinguished. To ensure the formation of V_2O_5 @-polypyrrole core-shell nanowires and avoid the excess displacement of the V_2O_5 cores by pyrrole, the molar ratio between V_2O_5 and pyrrole was set at around 1 : 0.66. Compared to the samples consisting solely of V_2O_5 nanowires or polypyrrole, the V_2O_5 @polypyrrole core-shell nanowires showed remarkable improvements in specific capacitance, rate performance and cycling stability. Nanofiber dispersions and nanofibrous membranes composed of randomly oriented polypyrrole nanofibers can be obtained by the reactive template methods in general, while new nanoarchitectures are rarely reported. By adding low-molecular-weight polyethylene glycol (PEG) into the reactive template of V_2O_5 nanofiber sol, bulk polypyrrole fibers composed of polypyrrole nanofibers assembled along the fiber long axis can be synthesized in a one-step manner.⁴⁶ This new nanoarchitecture of polypyrrole typically showed diameters in the range of 10–50 μm , while their lengths can reach several centimeters. The surface of the bulk polypyrrole fibers is not smooth, and is roughened by intensive creases and folds that can span over several tens of microns (Fig. 3D). The SEM image with higher magnification shows the detailed morphology of the creases and folds, revealing a hierarchical micro/nano-structure composed of stacking layers of polypyrrole nanofibers (Fig. 3E). Moreover, the micro-creases and folds formed by the mismatch between the layers are observed to be partially aligned with the long axis of the fiber, which can be attributed to the shear stress applied on the V_2O_5 nanofiber sol during the synthesis process. The surface nanostructure of the bulk polypyrrole fiber is shown in Fig. 3F, as a dense layer of interwoven polypyrrole nanofibers with an average diameter of 63 nm and a standard deviation of ± 7 nm can be observed. Compared to the polypyrrole nanofibers synthesized solely by V_2O_5 nanofiber sol and polypyrrole granules synthesized without adding the V_2O_5 sol, the bulk polypyrrole fibers showed enhanced electronic properties, *i.e.*, a larger band gap, a higher bipolaron density, and a higher degree of charge carrier delocalization, which can be ascribed to the strong hydrogen bonding and charge transfer interaction between the PEG molecules and polypyrrole.

Different from the reactive templates that drive the nanofiber formation through a redox mechanism, the nanofibrous templates that are able to protonate pyrrole monomers can also induce the formation of polypyrrole nanofibers. Pyrrole radicals can form by the protonation process, which can initiate the polymerization of pyrrole through a trimerization mechanism (Fig. 4). Polypyrrole nanofibers with unique helicity can be synthesized by using chiral amino acids as the template. For example, A. Xie *et al.* reported the use of left-handed and right-handed *N*-myristoyl-glutamic acids (L/R-MGA) to synthesize single-handed helical polypyrrole nanotubes with diameters of around 80 nm in an ethanol/water solution.⁴⁷ Polypyrrole nanotubes with either a left-handed helical structure or a right-handed helical structure can be obtained by using L-MGA or R-

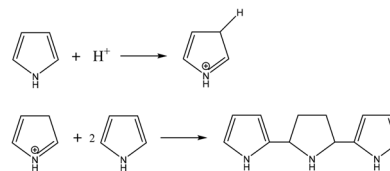
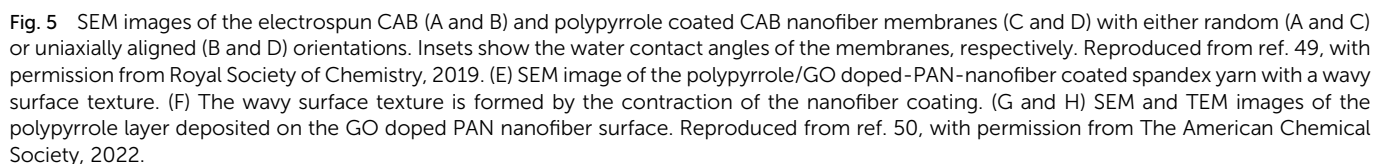


Fig. 4 Protonation and trimerization processes of pyrrole in acids.

MGA, respectively. Helical polypyrrole nanofibers with diameters of around 40 nm can also be obtained by replacing the MGA templates with a gel dispersion of *N*-myristoyl-L-diglutamic acid (L-MDGA). When being used as the electromagnetic interference (EMI) shielding materials, the helical nanostructured polypyrrole showed both high shielding efficiency and wide absorbing frequency range, while the highest absorption width of 7.04 GHz was obtained for the helical polypyrrole nanofibers in the frequency range between 10.6 and 17.64 GHz. By adding different acids into the polymerization solution of the helical polypyrrole nanotubes, M. Sun *et al.* were able to synthesize hollow polypyrrole nanofibers with tunable microstructures, which can be used for EMI shielding.⁴⁸

On the other hand, electrospun nanofiber mats can also be used as either non-reactive or reactive templates to synthesize polypyrrole nanofibers. For example, R. Elashnikov *et al.* prepared polypyrrole nanofiber membranes for cell culture and stimulation by using an electrospun cellulose acetate butyrate (CAB) nanofiber mat as the substrate;⁴⁹ SEM images of the as-obtained CAB nanofiber mat are shown in Fig. 5A and B. Polypyrrole was then deposited onto the CAB nanofiber mat by chronopotentiometry. Both randomly oriented and uniaxially aligned polypyrrole nanofiber membranes can be obtained from the electrospinning process, by using either a single plate or parallel plates as the collector (Fig. 5C and D). The as-obtained polypyrrole nanofiber membranes can significantly promote the growth of the human neuroblastoma cells (SH-SY5Y) when electrical stimulation is applied, resulting in longer neurites with more branches. Compared to the randomly oriented nanofiber mat, the uniaxially aligned nanofiber mat can induce directional neurite growth, resulting in a higher average length for the as-grown neurites. Y. Zhou *et al.* reported the fabrication of a high-performance yarn-based strain sensor by using the polypyrrole nanofiber composites.⁵⁰ According to their method, a spandex yarn was pre-stretched and fixed in between two twisting rollers connected to a winding device. The graphene oxide (GO) doped polyacrylonitrile (PAN) nanofibers were then electrospun onto the spandex yarn and covered its surface. Subsequently the nanofiber coated yarn was twisted by the twisting rollers to entangle the nanofibers and the spandex yarn. After removing the stretching load, the nanofibers would contract to form a uniform wavy structure on the surface of the yarn, as shown in Fig. 5E and F. Polypyrrole was then deposited onto the surface of the PAN nanofiber-coated spandex yarn by a free radical polymerization process in the aqueous phase, forming a continuous conductive layer on the PAN nanofibers' surface, as shown in Fig. 5G and H. The as-obtained composite yarn was highly adaptable to the weaving process and showed





5. Structure–property correlations of the polypyrrole nanofibers

Electrical conductivity is considered as one of the most important properties for conducting polymers, *e.g.*, polypyrrole, since the main applications of conducting polymers, such as in supercapacitors, organic solar cells, bioactive scaffolds, conductive fillers, and sensors, require high electrical conductivity to realize advanced functions and performance. The correlations between the electrical conductivities of the polypyrrole nanofibers and the corresponding synthesis methods in the literature are summarized in Table 1. From Table 1, it can be seen that the electrical conductivities of the nanofibers differ significantly based on the synthesis conditions, including the methods, morphologies, molar ratios, and dopants. By carefully investigating the detailed information and parameters in the literature, a few general rules can be deduced: (i) the electrical conductivity of the as-prepared nanofibers increases with the weight fraction of polypyrrole, especially for the electrospun nanofibers; (ii) the electrical conductivity of the polypyrrole nanofibers increases as the density of polypyrrole increases, *e.g.*, the conductivity of the solid polypyrrole nanofibers is generally higher than that of the hollow nanofibers or coatings; (iii) the conductivity of the as-prepared nanofibers depends on their composition, and the presence of other conductive components can enhance the electrical conductivity; (iv) organic dopants, such as benzenesulfonate anions, can induce higher electrical conductivity, which can be attributed to the electrostatic shielding effect originating from their large sizes; (v) protonation in the acidic solution may enhance the reactivity of pyrrole, resulting in the formation of polypyrrole with higher electrical conductivity. The unique nanofibrous morphology may provide anisotropic physical properties, larger surface area, and assemblability, which may significantly improve the efficiency of functionalization and nanoscale self-assembly. By taking advantage of the unique doping-induced properties of polypyrrole, *e.g.*, conductivity, structural change, and stimuli-responsiveness, important applications of polypyrrole nanofibers may be found across a wide range of research disciplines.

Table 1 Summary of the compositions, morphologies, and conductivities of the polypyrrole (PPy) nanofibers prepared by using different synthesis methods

Name	Synthesis method	PPy morphology	Molar ratio of Py to oxidant	Dopant	Conductivity	Testing method
PPy/chitosan/collagen nanofiber scaffold (15 wt% PPy) ⁴	Electrospinning	Blended in the electro-spun nanofibers; $d_{avg} = 83.7$ nm	0.95	Dioctyl sulfosuccinate, SO_4^{2-}	$\sim 164 \times 10^{-3} \text{ S m}^{-1}$	4-Probe, sheet form
PPy nanofibers ⁸	Interfacial polymerization	Nanofiber bundles and networks; diameter range: 75–180 nm	7.27 (0.3 for $FeCl_3$)	HCl , $FeCl_3$, ^a <i>p</i> -TSA, CSA, PSSA	$^{b}25 \times 10^{-4}$, 21×10^{-4} , 6×10^{-2} , 4×10^{-2} , $20 \times 10^{-4} \text{ S cm}^{-1}$	4-Probe, pressed pellet
PPy nanoribbons and nanowires ⁹	Reactive templating	Curved nanowire or nanoribbon networks; diameter: 20–65 nm	1	Br^- , SO_4^{2-}	$7.3 \times 10^{-3} \text{ S cm}^{-1}$	4-Probe, compressed pellet (10 MPa)
PPy/PVDF electrospun nanofibers ²⁰	Electrospinning	Blended in the electro-spun nanofibers; $d_{avg} = 325$ nm	0.5	$FeCl_3$	$\sim 1.6 \times 10^{-4} \text{ S cm}^{-1}$	LCR meter, nanofiber membrane
MnO_2 @PAA/PPy core-shell composite fibers ²²	Reactive templating	Granular coating, 50 nm thick	—	$FeCl_3$	$1.19 \times 10^{-3} \text{ S m}^{-1}$	2-Probe, fiber mat
PPy/PEO nanofibers ²⁶	Electrospinning	Blended in the electro-spun nanofibers; diameter range: 200–300 nm	—	Organic acids	4.9×10^{-8} to $1.2 \times 10^{-5} \text{ S cm}^{-1}$, depending on the PPy content	2-Probe, fiber mat
PPy/PVP nanofibers ²⁷	Electrospinning	Blended in the electro-spun nanofibers; $d_{avg} = 833$ nm	—	AQSA, $FeCl_3$	$5.22 \times 10^{-1} \text{ S cm}^{-1}$	4-Probe, nano-fiber mat
PPy nanofibers ³⁷	Interfacial polymerization	Nanofiber bundles	1	<i>p</i> -TSA	$3.3 \times 10^{-1} \text{ S cm}^{-1}$	4-Probe, pressed pellet
PPy nanofibers ³⁷	Interfacial polymerization	Nanofiber bundles	1	HCl	$9.7 \times 10^{-2} \text{ S cm}^{-1}$	4-Probe, pressed pellet
Poly(pyrrole nanofibers) ³⁹	Reactive templating	Nonwoven mat of nanofibers	4.3	SO_4^{2-} , Cl^- , Br^-	$\sim 100 \text{ S cm}^{-1}$	Compressed pellet
PPy nanoclip ⁴⁰	Reactive templating	2D nanoclips	4	SO_4^{2-} , Cl^- , Br^-	$2\text{--}5 \text{ S cm}^{-1}$	4-Probe, compressed pellet
V_2O_5 @PPy core-shell nanowires ⁴⁵	Reactive templating	Nanocoating, 15 nm thick	0.65	DBS [−] , SO_4^{2-}	—	—
PPy nanofiber assemblies ⁴⁶	Reactive templating	Uniaxially aligned bulk nanofiber assembly; $d_{avg} = 63$ nm	3.27	SO_4^{2-} , Cl^-	1.49 S m^{-1}	4-Probe, pressed pellet
PPy coated cellulose acetate butyrate nanofibers ⁴⁹	Non-reactive templating	Granular coating, 600 nm thick	—	$H_2PO_4^-$	$2.4 \times 10^{-5} \text{ S cm}^{-1}$	2-Probe, nano-fiber mat
Helical PPy nanofibers ⁵³	Non-reactive templating	Nanofibers with helical architectures; $d_{avg} = 113$ nm	1	SO_4^{2-}	1.83 S cm^{-1}	—
ANF-PVA-PPy hydrogel ⁵⁴	Non-reactive templating	Granular coating	—	$FeCl_3$	$\sim 80 \text{ S cm}^{-1}$	2-Probe, cuboid hydrogel

^a Abbreviations: d_{avg} (average diameter), *p*-TSA (*p*-toluene sulfonic acid), CSA (camphor sulfonic acid), PSSA (polystyrene sulfonic acid), PVDF (polyvinylidene fluoride), LCR (inductance, capacitance, and resistance), PAA (polyacrylic acid), PEO (polyethylene oxide), PVP (polyvinyl pyrrolidone), AQSA (anthraquinone-2-sulfonic acid sodium salt), DBS (dodecylbenzenesulfonate), ANF (aramid nanofibers), PVA (polyvinyl alcohol). ^b The conductivities of PPV nanofibers doped with HCl, FeCl₃, *p*-TSA, CSA, and PSSA, respectively.

6.1 Biomedical applications

6.2 Energy storage and conversion

6.3 Multifunctional actuators

(electrochemical actuation) are the pros of the PPy based actuators, but challenges still exist, *e.g.*, how to simultaneously realize flexibility, durability, fast response and large stroke length. Hybridization of PPy with other carbon-based materials such as graphene and carbon nanotubes shows noticeable promise for constructing high-performance actuators.⁶⁷ Moreover, the incorporation of PPy into the wafer-scale nanoporous single-crystalline silicon membrane provided new perspectives for on-chip actuators with outstanding voltage-strain coupling and small operation voltages.⁶⁸

6.4 Sensors

Numerous studies have been conducted to develop PPy-based sensors, such as electrochemical, gas, and bio-sensors, since PPy is highly promising for commercial sensory applications as a result of its solution-processability, stimuli-responsiveness, and good environmental stability.⁶⁹ For biosensors, the analysis of selected DNA sequences and mutated genes associated with human disease is of significant importance. PPy may serve as a facile, high-performance and inexpensive functional material for fulfilling this goal. For example, a polypyrrole-based DNA chip has been designed and applied to the analysis of genomic DNA originating from cell lines and human colorectal samples, while practical features, such as low cost, high accuracy, and adaptability, can be realized.⁷⁰ Because of its high conductivity and electrochemical activity, PPy has been integrated with different substrates for collecting acute signals such as from genes,⁷¹ ammonia gas, thermal heating, NIR light,⁷² and volatile organic compounds.⁷³

7. Conclusions

Polypyrrole nanofibers with different compositions, nanoarchitectures and combinations can be synthesized by using electrospinning, interfacial polymerization, and solution-based template methods. For the electrospinning method, the electrospun nanofiber mats can be utilized as the substrate for polypyrrole deposition and as the templates to direct nanofiber formation. A variety of polypyrrole nanofiber-based nanoarchitectures can be facilely fabricated by the electrospinning technique, including core-shell nanofibers, hollow nanofibers, and uniaxially aligned nanofibers. Moreover, advanced fiber/nanofiber-based composites and devices can be fabricated by coating polypyrrole on the surface of the composite nanofibers prepared by the electrospinning of a multi-component dope solution or co-electrospinning of different dope solutions. Enhanced electrical conductivity, charge storage capacity, photothermal properties and biocompatibility can be provided by the polypyrrole coating layer, while enhanced mechanical properties, processability, and wearability can be provided by the electrospun nanofiber substrates. Given the diversity in composition, nanoarchitecture and combination, the polypyrrole nanofiber-based nanoarchitectonics show promising performance in a wide range of applications, *e.g.*, cell culture, neural stimulation, photothermal therapy, electronic skins, and strain sensors.

Compared to the electrospinning technique, interfacial polymerization can generate polypyrrole nanofibers without the presence of a template. It is speculated that a pH-dependent adsorption/desorption process at the oil–water interface may drive the formation of the nanofibers. As the polymer chains of polypyrrole are protonated, de-coiled, and desorbed into bulk water at the oil–water interface, the as-obtained polypyrrole may take up more extended and linear conformations, forming a stable dispersion in water due to the electrostatic repulsion. Consequently, curved polypyrrole nanofibers with small diameters are generally obtained by interfacial polymerization. On the other hand, the conformation of the polymer chains can be effectively tuned in the presence of the oil–water interface, which may enhance the stacking order of the polymer chains, resulting in a higher crystallinity and conductivity.

Compared to the interfacial polymerization and electrospinning technique, the synthesis of polypyrrole nanofibers by using reactive templates does not require complicated spinning devices, post-spinning deposition, or tedious interfacial adsorption/desorption processes, and the synthesis process can be completed in a simple, fast, and environment-friendly manner. Different from the non-reactive templates, the reactive templates can initiate the polymerization of pyrrole through redox mechanisms, driving the formation of polypyrrole on their surface and forming polypyrrole that duplicates their nanofibrous morphologies. Polypyrrole nanofibers with versatile morphologies and nanoarchitectures can be obtained by using the reactive templates, including the highly curved nanofibers, long and straight nanofibers, nanotubes, core-shell nanofibers, nanofiber urchins, and hierarchical polypyrrole bulk fibers, which show a wide range of applications in supercapacitors, lithium-ion batteries, organic electronics, and EMI shielding.

Conflicts of interest

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