



Polarization effects have also been observed in NPLIN of *L*-histidine,<sup>26</sup> carbamazepine<sup>28</sup> and sulfathiazole.<sup>29</sup>

The mechanisms of NPLIN are not fully understood. Mechanisms have been proposed based on the optical Kerr effect and the response of solute clusters in solution to an electric field.<sup>16,17</sup> More recently, laser heating of nanoparticle impurities in the solution, resulting in the formation of vapour cavities, has been proposed as a mechanism that could account for a number of previously unexplained observations.<sup>20,36,37</sup> For more details see, for example, a recent review by Alexander and Camp.<sup>40</sup>

Polymorph and solvate control have been demonstrated by laser-induced nucleation (LIN) using different light sources. Using focused femtosecond laser pulses, where sufficient power can be delivered to the system for optical breakdown and laser-induced cavitation, selective formation of metastable polymorphs of indomethacin and aspirin were demonstrated by Maruyama and co-workers.<sup>41,42</sup> Wang *et al.* also succeeded in increasing the percentage of metastable forms II and III of paracetamol from less than 10% to more than 40% using focused femtosecond pulses.<sup>43</sup> A versatile technique is laser-trapping crystallization, as developed by Masuhara, Sugiyama and co-workers, in which a continuous-wave (CW) laser beam is focused at a solution–air interface.<sup>44</sup> Crystallization occurs *via* the localization of solute clusters in the focal region, leading to an increased supersaturation. In glycine/D<sub>2</sub>O solutions, the favoured polymorph was dependent on the concentration, temperature, laser power, and polarization.<sup>14,45</sup> The polarization switching was explained in terms of the efficiency of trapping of  $\alpha$  and  $\gamma$ -glycine precursor clusters with CP and LP light, as well as conversion from  $\alpha$  to  $\gamma$  at high concentrations. More recently, control of hydrate formation in *L*-phenylalanine has been achieved using trapping with a CW-laser beam, or with trains of femtosecond pulses.<sup>15,46,47</sup>

The aim of the present work was to investigate hydrate formation in a simple inorganic salt system using NPLIN. Sodium bromide was selected as a good candidate: an aqueous supersaturated solution under ambient pressure conditions can yield either dihydrate (DH) or anhydrous (AH) crystals. NPLIN was tested at different points on the solute–solvent phase diagram and the resulting crystals were characterised. The results were compared to mechanical shock-induced nucleation (MSIN), sonocrystallization and laser trapping. The results show the ability to control hydrate crystal formation by laser-induced nucleation, and reveal new features of the underlying mechanism.

## 2. Experimental methods

Anhydrous sodium bromide (Sigma Aldrich, ReagentPlus,  $\geq 99\%$ ) and ultrapure water (18.2 M $\Omega$  cm) were used to prepare concentrated solutions of aqueous sodium bromide. In the present work, concentrations (*C*) are expressed as moles of anhydrous salt per kg of solvent. Stock solutions with concentrations of 10.0, 10.9 and 11.5 mol kg<sup>-1</sup> were

prepared and held in a temperature-controlled oven ( $T = 75$ – $95$  °C) for two days to ensure dissolution. The warm solutions were then transferred to pre-heated glass sample vials ( $\sim 5$  mL). The vials were sealed and reheated to dissolve any spontaneously formed crystals. The solubility of sodium bromide was determined by fitting tabulated experimental data, and the resulting saturation concentrations ( $C_{\text{sat}}$ ) were used to calculate supersaturations ( $S = C/C_{\text{sat}}$ ).<sup>48,49</sup> The solution conditions are summarized in Fig. 1 (see ESI† for more details). Supersaturation was in the range 1.01 to 1.29. In terms of Fig. 1, the height of the point above the solubility curve, either the AH curve or the DH curve, is a measure of supersaturation with respect to that solid form. DH is the most stable crystal form below the transition temperature of 51 °C.<sup>48,49</sup> The dashed curve in Fig. 1 represents extrapolation of the AH solubility curve data to lower temperatures. We justify this extrapolation based on the analogous sodium-acetate hydrate system, for which solubility data exist in this region below the transition temperature (58 °C).<sup>50</sup>

The occurrence of NPLIN in sodium bromide solutions was established by counting the number of samples nucleated after exposure to single laser pulses. A set of sample solutions at 11.5 mol kg<sup>-1</sup> were cooled to either 60, 55, 50 or 45 °C in a water bath ( $\pm 1$  °C). These temperatures were chosen to explore the transition region around 51 °C while minimising spontaneous nucleation. After cooling for one hour, each sample was checked for spontaneously formed crystals. Samples which had not nucleated were

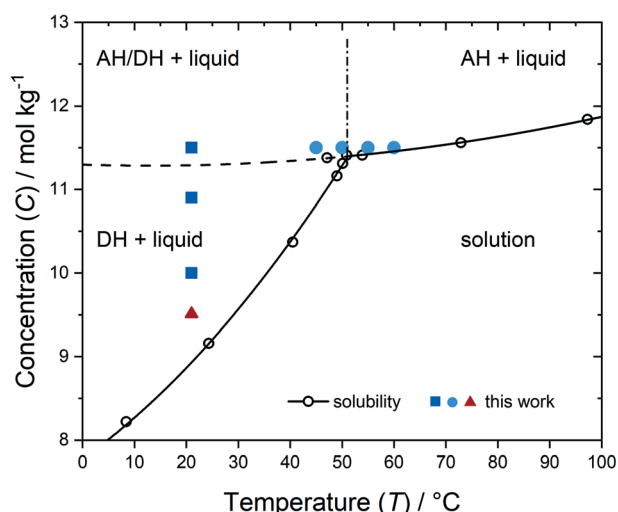


Fig. 1 Phase diagram based on solubility data for NaBr + H<sub>2</sub>O (open circles).<sup>49</sup> The most-stable equilibrium solid form in contact with the liquid solution is indicated on the plot (AH = anhydrous; DH = dihydrate). The solid curves represent quadratic model fits to the solubility data. The dashed curve represents an extrapolation to low *T* of the fit of AH data above the transition temperature (51 °C). The vertical line (dot-dash) is simply to guide the eye at the transition temperature. The experimental conditions in the present work are represented as solid symbols. Note that the laser trapping experiments (solid triangle) were conducted with NaBr + D<sub>2</sub>O, but the phase diagram is qualitatively similar to the one shown.



placed in a temperature-controlled sample holder and exposed to a single pulse of laser light (532 nm) with an incident energy density of  $1.0 \text{ J cm}^{-2}$  (see ESI† for details). After irradiation, samples were returned to the water bath for an hour, before being checked for crystals. Any samples that had not nucleated were cooled to room temperature and nucleated by mechanical shock (hitting firmly on the bench top) before re-use. Control experiments at each temperature were carried out, where the vials were treated in the same way but not exposed to the laser pulse.

The behaviour of sodium bromide solutions at room temperature ( $21 \text{ }^\circ\text{C}$ ) was investigated by cooling sample solutions slowly overnight. Each vial was exposed to a single pulse of laser light (1064 nm) with an incident energy density of  $0.6 \text{ J cm}^{-2}$  (see ESI† for details). In this case, near-infrared laser light was used to compare against visible light. We found that the wavelength of light used (532 or 1064 nm) made no discernible difference to the observations. A digital camera with a macro-zoom lens was used to image the crystal growth at the bottom of the vial. Only samples that nucleated within one hour were counted.

Nucleation of samples by mechanical shock, sonocrystallization, and laser trapping (for NaBr in  $\text{D}_2\text{O}$ ) was investigated at  $21 \text{ }^\circ\text{C}$ . For investigation of MSIN, sample vials were (by hand) struck firmly at the base against a flat metal surface. For investigation of sonocrystallization, samples were immersed for approximately one second in an ultrasonic bath (Elmasonic S-30-H, 37 kHz, nominal power 80 W) then left at room temperature, and checked periodically for crystals. The procedure was repeated each day for any samples which had not nucleated (for a maximum of 5 days). Control experiments were also carried out, in which samples were treated in the same way but the ultrasonic bath was not turned on.

For nucleation by laser trapping, a CW laser was used (1064 nm, Laser Quantum Ventus).  $\text{D}_2\text{O}$  was used as the solvent to minimise heating over time by the laser at this wavelength.<sup>51,52</sup> The phase diagram for NaBr in  $\text{D}_2\text{O}$  is very similar to that in  $\text{H}_2\text{O}$ , with a slightly lower transition temperature of  $47 \text{ }^\circ\text{C}$ .<sup>53</sup> A metastable solution was prepared by dissolving the salt at  $40 \text{ }^\circ\text{C}$  and adding  $\text{D}_2\text{O}$  dropwise until the solution was stable on cooling to room temperature ( $21 \text{ }^\circ\text{C}$ ). The resulting concentration was  $9.51 \text{ mol kg}^{-1}$ , which corresponds to a supersaturation of  $S = 1.08$  (DH) =  $0.88$  (AH), *i.e.*, the solution is undersaturated with respect to AH.<sup>53</sup> Droplets of solution ( $<0.5 \text{ } \mu\text{L}$ ) were arranged on top of a glass coverslip that was bonded within a container to inhibit evaporation. The laser was focused at the solution–air interface from underneath the droplet using a long-working distance objective (Mitutoyo 50 $\times$ , NA = 0.42). The diameter of the beam at the focal spot was approximately  $1.5 \text{ } \mu\text{m}$ . The average power of the beam was controlled to be in the range  $1.0\text{--}1.2 \text{ W}$ , as measured after the objective with no sample in place. Crystal formation was recorded through the same objective, using a digital camera fitted with an optical filter to block the laser wavelength.

## 3. Results

### 3.1 NPLIN and spontaneous nucleation

Two morphologies of sodium bromide crystals were observed: (1) clear, cubic crystals that grow slowly to a maximum length of a few millimetres (Fig. 2a); (2) plate-like crystals that grow either very rapidly, filling the vial within a few seconds as a solid translucent mass (Fig. 2b), or growing more slowly into larger individual sheets, leaving some remaining solution (Fig. 2c). The plate-like crystals were stable at room temperature and were the type formed by spontaneous nucleation or by seeding with AH sodium bromide crystals (see ESI† for details). The crystal structures of the cubic and plate-like morphologies were determined by single-crystal X-ray diffraction and assigned to anhydrous and dihydrate sodium bromide, respectively (see ESI† for details). Solutions that had started to grow cubic crystals at room temperature could also nucleate plate-like crystals, either spontaneously, by seeding, or by being knocked: this result is consistent with the phase diagram shown in Fig. 1. The phase diagram normally represents the attainment of an equilibrium form. If AH and DH are both present in the same vessel below  $51 \text{ }^\circ\text{C}$ , one would expect to see AH transform to DH over time as equilibrium is achieved. How fast this happens (kinetics) depends on solution parameters such as temperature and the size of crystals.

Table 1 summarizes the number of vials nucleated and the type of crystals formed at different temperatures. At  $50 \text{ }^\circ\text{C}$  and above, all crystals formed were AH, which is the expected solid form above  $51 \text{ }^\circ\text{C}$  (Fig. 1). Between one and 20 AH crystals formed in each vial. The lack of nucleation in the corresponding control experiments means we have confidence that these crystals were nucleated by the laser. At  $45 \text{ }^\circ\text{C}$ , some DH crystals were formed in the laser experiments: one or two plate-like sheets appeared in each vial. Since DH crystals were also formed in the control experiments at this temperature, it is not certain if these crystals were laser-induced or not.

In contrast to the results at  $45\text{--}60 \text{ }^\circ\text{C}$ , when samples were cooled to room temperature ( $21 \text{ }^\circ\text{C}$ ), it was found that many samples nucleated spontaneously due to the high supersaturations being accessed ( $S = 1.12\text{--}1.29$ ). Spontaneous nucleation resulted in the formation of DH crystals in all cases except one. Increasing the temperature at which the samples were dissolved significantly reduced the occurrence of spontaneous nucleation. When samples were dissolved at  $75 \text{ }^\circ\text{C}$ , spontaneous nucleation occurred in 34/43 vials on cooling, but at  $90 \text{ }^\circ\text{C}$  the frequency of spontaneous nucleation reduced to 7/52 vials. It is possible that the higher temperature ensures dissolution of all solids and aids in degassing the solution, making the samples less liable to nucleation when being handled.

Table 2 shows the number of samples nucleated at  $21 \text{ }^\circ\text{C}$  after exposure to the laser. Remarkably, only 2/23 samples formed DH after nucleation. In particular, the majority (21/22) of  $11.5 \text{ mol kg}^{-1}$  samples formed AH sodium bromide.





**Fig. 2** Images of the different crystal morphologies formed from aqueous supersaturated solutions of sodium bromide: (a) small colourless cubic crystals (AH) (scale bar = 2 mm), (b) a translucent mass consisting of flat plates of crystals (DH) that fill the original solution (scale bar = 5 mm), (c) transparent flat plates (DH) leaving a larger fraction of solution at the top (scale bar = 5 mm). Solution conditions:  $C = 11.5 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.29$  (DH) or  $1.02$  (AH).

The AH crystals grew very slowly, and were visible by eye only after 20–30 minutes. However, closer examination of the recorded images showed crystallites falling to the bottom of the vial 10–100 s after the laser pulse. Fig. 3 shows the growth of a typical group of laser-induced AH crystals following exposure to a laser pulse.

### 3.2 Nucleation by mechanical shock and sonocrystallization

Samples that were nucleated deliberately by mechanical shock ( $T = 21 \text{ }^\circ\text{C}$ ) always formed DH crystals. One hit was usually sufficient, although in some cases two or three hits were required. Two different growth behaviours were observed. (1) In some samples, nucleation occurred in localised regions of the solution, often near the solution–air interface (Fig. 4), possibly where liquid had splashed up the walls of the vial. Plate-like crystals grew outwards from these regions until the entire solution was filled with a translucent mass. (2) In other samples, a larger number of nucleation centres were observed throughout the solution (Fig. 5). Smaller plate-like crystals formed and fell to the bottom of the vial: these crystals were smaller and grew more quickly than case (1).

Table 3 shows the number of samples that nucleated after immersion in the ultrasonic bath, compared to the control samples. Only DH crystals were observed, except for ultrasonication at concentrations above saturation for AH ( $11.5 \text{ mol kg}^{-1}$ ) where approximately half of the samples nucleated produced AH crystals.

### 3.3 Laser-trapping nucleation

Only AH crystals were observed in the laser trapping of solutions of NaBr in  $\text{D}_2\text{O}$ . In each of several droplets, a single cubic crystal was observed growing at or near the focal point within 10 s to 4 minutes of exposure to the beam. The growth rate of the crystals could be controlled by adjusting the laser power: at the maximum power (1.2 W) steady growth was maintained; while at lower power (1.0 W) slower growth occurred, and the crystal was less stable in the trap. The crystal could be translated along the interface while trapped as it was pinned by the laser to the interface, but moving the focal volume into the solution resulted in loss of trapping. Trapped crystals were observed to grow out from the focus of the laser, as shown in Fig. 5 (see Video S5 in ESI†). When nucleation was first observed, the crystals produced ( $<1 \text{ }\mu\text{m}$ )

**Table 1** NPLIN of NaBr crystals in the range of temperatures 45–60 °C. The nominal supersaturation of the solution ( $11.5 \text{ mol kg}^{-1}$ ) with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of vials that nucleated after exposure to a single laser pulse (532 nm), in comparison to a control experiment with no laser. Samples that nucleated during the initial hour of cooling prior to exposure were excluded, hence the variation in the total number of samples tested

Temperature ( $T$ )/ $^\circ\text{C}$	Supersaturation ( $S$ )	Control			With laser		
		Total tested	Nucleated		Total tested	Nucleated	
			DH	AH		DH	AH
60	1.004 (AH)	7	0	0	10	0	3
55	1.01 (AH)	10	0	0	10	0	4
50	1.02 (DH)	20	0	0	19	0	8
	1.01 (AH)						
45	1.06 (DH)	10	2	1	5	2	1
	1.01 (AH)						





**Table 2** NPLIN of NaBr crystals at 21 °C. The supersaturation with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of vials of solution that nucleated (out of the total tested) after exposure to a single laser pulse (1064 nm). Samples that nucleated spontaneously during the slow cooling overnight, prior to testing, were not counted

Concentration (C)/mol kg <sup>-1</sup>	Supersaturation (S)	Total tested	Nucleated with laser	
			DH	AH
11.5	1.29 (DH)	30	1	21
	1.02 (AH)			
10.9	1.22 (DH)	6	1	0
	0.97 (AH)			
10.0	1.12 (DH)	4	0	0
	0.89 (AH)			

tended to reorientate or spin in the trap due to asymmetry in the shape of the crystal. After some growth (>3 μm) a more-stable trapping configuration was obtained, with the crystal being pinned at an edge or corner, and sometimes jumping from trapping at one corner to another.

## 4. Discussion

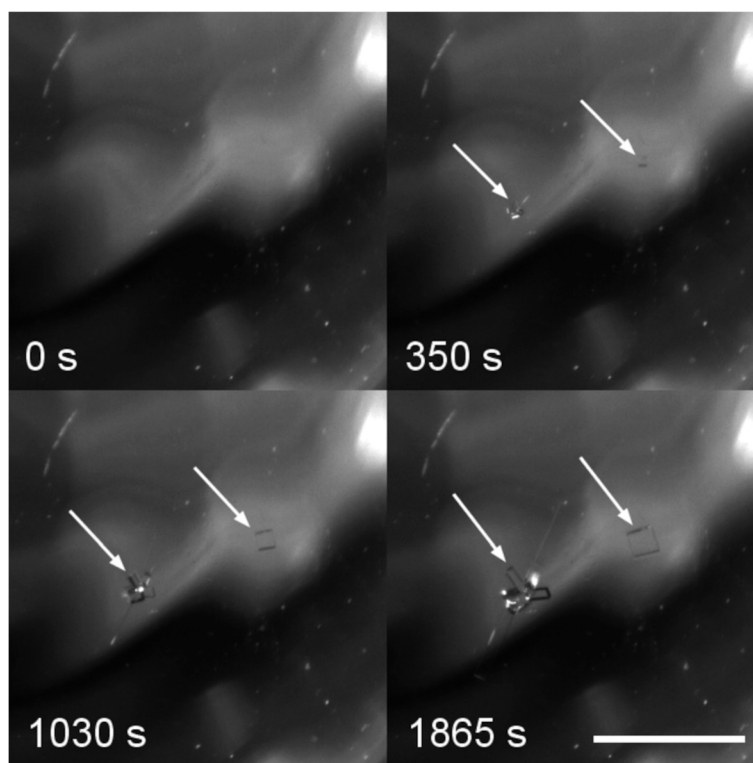
A summary of the observed preferences for hydrate formation using the different methods is given in Table 4. We can group the methods into three categories: (1) strong

preference for DH crystals (spontaneous and mechanical shock); (2) both DH and AH can be formed (sonocrystallization); (3) strong preference for AH crystals (NPLIN, laser trapping). In the remainder of the discussion, we use the results to extract features of the mechanisms involved.

### 4.1 Laser-trapping nucleation

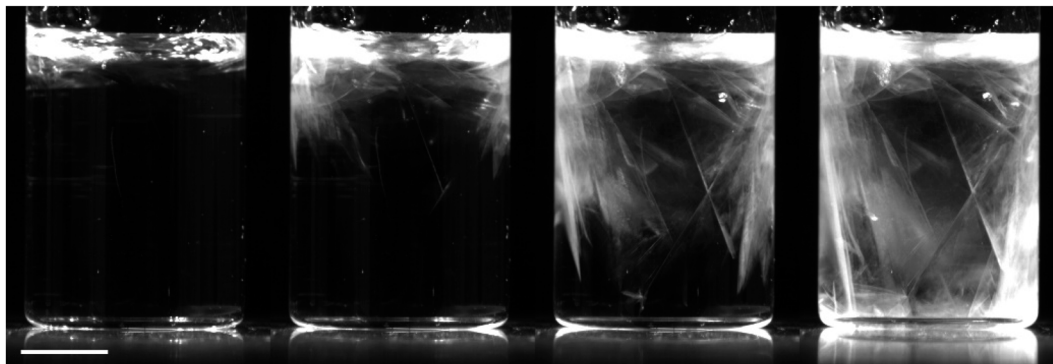
The preference for nucleation of AH *via* laser-trapping is remarkable because the starting solution at  $T = 21$  °C was supersaturated with respect to DH ( $S = 1.08$ ) but undersaturated with respect to AH ( $S = 0.88$ ). The ability to nucleate crystals in undersaturated solutions was first reported by Sugiyama and co-workers for glycine in D<sub>2</sub>O.<sup>54</sup> The CW laser at 1064 nm is expected to cause localised heating due to absorption by the water (vibrational overtones), but the use of D<sub>2</sub>O rather than H<sub>2</sub>O limits this effect to approximately 2 K W<sup>-1</sup> of laser power.<sup>51,52</sup> Therefore, the observed preference cannot be explained simply by elevation of solution above the transition temperature of 47 °C, where AH becomes the preferred solid form in D<sub>2</sub>O (Fig. 1).

Yuyama *et al.* have conducted detailed studies on laser-trapping nucleation of the hydrate system L-phenylalanine (L-Phe) in water.<sup>15,46,51,55–57</sup> Monohydrate (MH) L-Phe is the stable crystal form at room-temperature, and anhydrous L-Phe (AH) is the stable form above 36 °C. For L-Phe in H<sub>2</sub>O a



**Fig. 3** Images of crystal growth of NaBr (AH) from an aqueous supersaturated solution following NPLIN. Solution conditions:  $C = 11.5$  mol kg<sup>-1</sup>,  $T = 21$  °C,  $S = 1.29$  (DH) or 1.02 (AH). The crystals were imaged through the bottom of the vial and are shown between 0 and 31 minutes after the laser pulse (1064 nm). Crystals were first visible approximately 25 s after the laser pulse. Scale bar represents 2 mm.





**Fig. 4** Nucleation by mechanical shock case (1) where nucleation occurred near the solution-air interface. Images of crystal growth of NaBr (DH) from an aqueous supersaturated solution following nucleation by mechanical shock, from left to right: 3.4, 9, 14, and 19 s after nucleation. Solution conditions:  $C = 10.9 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.22$  (DH) or 0.97 (AH). Scale bar represents 5 mm.

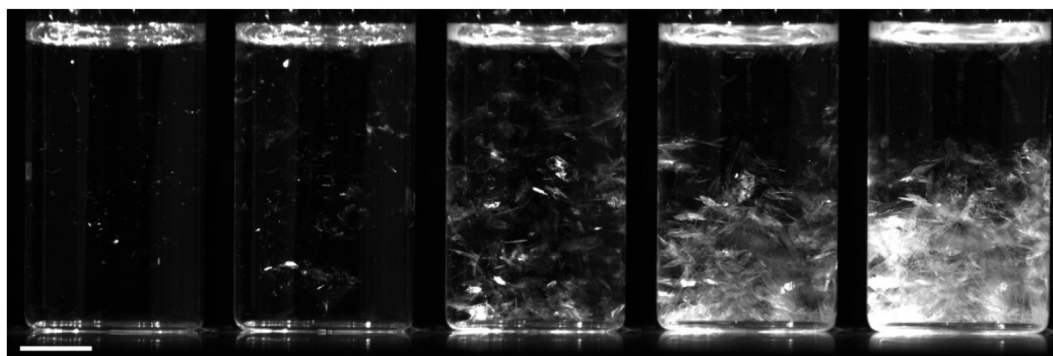
plate-like AH crystal was formed at the focal point, which grew out from its centre in all directions.<sup>46</sup> For L-Phe in  $\text{D}_2\text{O}$ , needle-shaped MH crystals nucleated, but only far away ( $>500 \text{ }\mu\text{m}$ ) from the laser focal volume.<sup>51</sup> It was considered that in  $\text{D}_2\text{O}$ , the interaction between laser and solute at the focal point is much stronger than in  $\text{H}_2\text{O}$ , to the extent that nucleation is hindered there. Nucleation may then occur outside the focal area, where the local structure of the solution favours MH. In the present work, we see growth of a cubic crystal outwards from a trapped corner (Fig. 6), which is likely due to the cubic morphology of AH NaBr *versus* the plate-like morphology of AH L-Phe.

Cheng *et al.* studied laser-trapping nucleation of KCl in  $\text{D}_2\text{O}$ .<sup>52</sup> At low laser powers (0.4 W) the crystal morphology was needle-shaped, but at higher laser powers (1.2 W) a cubic morphology was obtained, similar to the present work. KCl is not known to form hydrates under ambient conditions, and the needle morphology was attributed to growth at high supersaturation. The high concentration was possibly built up over tens of minutes of exposure to the laser prior to nucleation.

Based on previous mechanisms described by Sugiyama and co-workers, we consider that for aqueous NaBr solutions,

the laser acts to build-up concentration of solute around the focal volume.<sup>54</sup> By the time nucleation happens, the local concentration is above the AH solubility line. This can be thought of as an almost vertical transition on the phase diagram, from the bulk concentration (red triangle, Fig. 1) to above the AH solubility curve (dashed curve, Fig. 1). In the case of NaBr in  $\text{D}_2\text{O}$ , this transition would require an increase from  $9.51 \text{ mol kg}^{-1}$  to above  $10.8 \text{ mol kg}^{-1}$ , which is a 14% increase in local concentration. The observation that AH nucleates instead of DH is consistent with Ostwald's rule of stages, which states that the more thermodynamically unstable form will tend to nucleate first. Because the trapping method only creates one localised crystal, there are no DH crystals to promote transition to the more stable form.

We note that  $\text{D}_2\text{O}$  was used as solvent in the laser-trapping experiments only to prevent heating during long-term exposure to the 1064 nm CW trapping beam. A similar experiment could be conducted with NaBr in  $\text{H}_2\text{O}$  using a CW beam at a different wavelength (*e.g.*, 532 nm) where heating of the solvent would be minimal. We expect this would produce results identical to the present work, *i.e.*, nucleation of AH crystals.



**Fig. 5** Nucleation by mechanical shock case (2) where nuclei are initially formed in the bulk of the solution. Images of crystal growth of NaBr (DH) from an aqueous supersaturated solution following nucleation by mechanical shock, from left to right: 0.5, 2.8, 10, 17, and 34 s after nucleation. Solution conditions:  $C = 10.0 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.12$  (DH) or 0.89 (AH). Scale bar represents 5 mm.



**Table 3** Nucleation by sonocrystallization. The nominal supersaturation with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of aqueous sodium bromide samples that nucleated after immersion in an ultrasonic bath for approximately one second, with a maximum of five consecutive attempts, at 21 °C

Concentration (C)/mol kg <sup>-1</sup>	Supersaturation (S)	Control		With ultrasound			
		Total tested	Nucleated		Total tested	Nucleated	
			DH	AH		DH	AH
11.5	1.29 (DH)	21	7	0	26	13	10
	1.02 (AH)						
10.9	1.22 (DH)	19	11	0	16	12	0
	0.97 (AH)						
10.0	1.12 (DH)	7	0	0	11	4	0
	0.89 (AH)						

#### 4.2 Pulsed-laser NPLIN

The pulsed-laser NPLIN results show a strong preference for AH (90%). This preference can be rationalised in terms of the particle-heating mechanism for NPLIN.<sup>20,37,40</sup> The mechanism we describe here draws from the experiments of Soare *et al.* and corresponding simulations of Hidman *et al.*<sup>58,59</sup> In their work, crystal nucleation was induced following absorption of a focused laser pulse (~5 kJ cm<sup>-2</sup>) in a supersaturated solution, leading to thermocavitation. By contrast, in the present work the energy density ~1 J cm<sup>-2</sup> is substantially lower, and the absorbing medium is not the solution itself but nanoparticles in solution.

The mechanism is illustrated in Fig. 7. A rapid increase in temperature occurs due to absorption of the laser pulse by a solid nanoparticle (NP). Based on previous findings, the NP is likely to be a trace impurity in the solution, such as iron oxide or carbon.<sup>20,39</sup> The transient high temperature produces a layer of superheated layer (SHL) of liquid around the particle, leading to rapid vaporization and formation of an expanding cavity, *i.e.*, thermocavitation.<sup>60</sup> Vaporization continues from an interfacial layer (IL) at the expanding gas-liquid interface, giving rise to increased local supersaturation (Fig. 7c). The IL is a region of dehydration, which leads to the strong preference for AH while hindering formation of DH.

With reference to the phase diagram (Fig. 1) at 21 °C, at the highest experimental concentration (just above the AH solubility curve) 22/30 samples nucleated, of which 21 were AH. Below this line, only 1/10 samples nucleated, producing DH. This means that NPLIN operates in the region near or above the AH line. We should be cautious and note, however, that we only record events that result in successful growth of a crystal. It is possible that nuclei are formed but redissolve before they have grown sufficiently large to survive for us to observe the outcome.

#### 4.3 Sonocrystallization and mechanical shock

The use of sonocrystallization for nucleation and control of particle size is well established, and acoustic cavitation is considered to be the primary cause.<sup>61</sup> The cavitation that takes place is induced by shear from pressure waves

travelling through the liquid. We believe this causes colder and less-dehydrating nucleation events than NPLIN: at least under ultrasonic bath frequency and power conditions used in the present work. The observed branching between DH and AH neatly represents the position of the initial solution on the phase diagram: at experimental points below the AH solubility line only DH was observed; above this line we observed nucleation of almost equal numbers of AH and DH samples. We note that the exposure time to ultrasound in the present work was fixed at 1 s, in order to focus on primary nucleation. Longer exposure times would modify the ratio of AH to DH, since secondary nucleation will also become important.

The method we used for MSIN involves striking the base of the vial on a hard surface. A similar method is used in the common prank of beer tapping, where sharp knocking of a carbonated beer bottle releases a foam of bubbles.<sup>62</sup> The rapid acceleration of the vessel causes liquid shear and cavitation. We previously used MSIN to study nucleation of  $\alpha$  and  $\gamma$  polymorphs of glycine.<sup>30</sup> We found that the probability of nucleating  $\gamma$ -glycine crystals increased more sharply as a function of supersaturation using NPLIN compared to MSIN. We interpreted the results by suggesting that NPLIN accessed higher local supersaturations.<sup>30</sup> The results of the present work now suggest that the interface of the expanding cavity in NPLIN is also hotter, more dehydrating, which leads to the preference for AH crystals. Relative to NPLIN and sonocrystallization, we consider shock to cause colder cavitation events, and therefore we observe a stronger preference (100%) for DH.

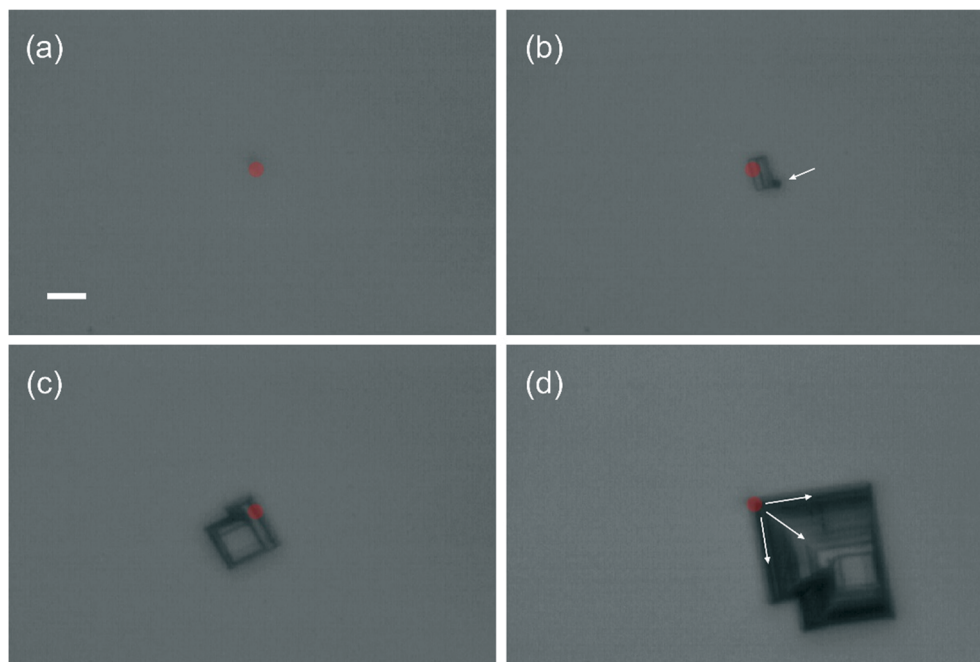
Finally, we comment on the observations of spontaneous nucleation. Spontaneous, homogeneous

**Table 4** Summary of preferences for NaBr crystal phases using different nucleation methods (DH = dihydrate; AH = anhydrous). For laser trapping, D<sub>2</sub>O was used as solvent; for all the other methods H<sub>2</sub>O was used

Nucleation method	NaBr preferred solid form
Spontaneous	DH (95%)
Mechanical shock	DH (100%)
Sonocrystallization	DH (74%)
NPLIN (pulsed laser)	AH (90%)
Laser trapping (CW laser)	AH (100%)





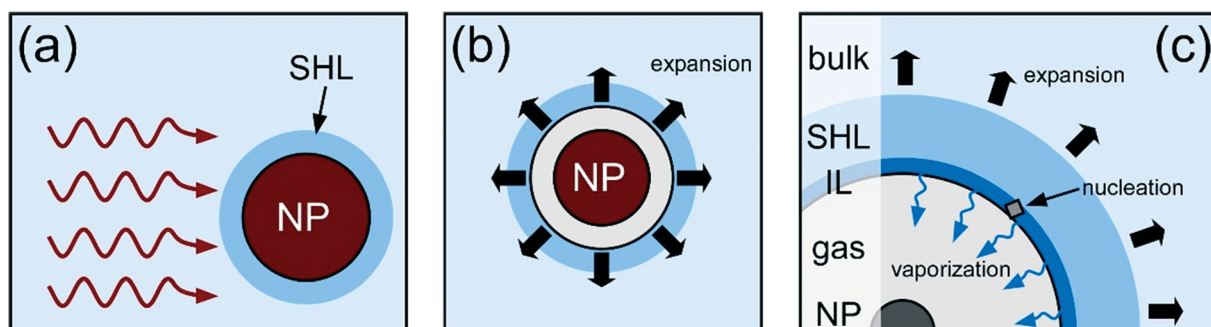


**Fig. 6** Laser-trapping nucleation and crystal growth of NaBr (AH) from an aqueous supersaturated solution in D<sub>2</sub>O. Solution conditions:  $C = 9.51 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.08$  (DH) or  $0.88$  (AH). Laser power was  $1.0 \text{ W}$  at  $1064 \text{ nm}$ . The scale bar represents  $10 \text{ }\mu\text{m}$ . The transparent red dot represents the location of the laser focus in the image plane (focal size not to scale). Images taken at times (a)  $11 \text{ s}$ , (b)  $64 \text{ s}$ , (c)  $78 \text{ s}$ , and (d)  $110 \text{ s}$  after laser was focussed at solution–air interface of droplet. The arrow in (b) shows a small crystallite attached to the main crystal. The arrow in (d) indicates the direction of growth of the crystal, outward from the laser focus, which can be followed by the crystalline steps visible on the crystal. See ESI† (Video S5) for a movie of the nucleation and trapping.

primary nucleation is highly unlikely under normal laboratory conditions.<sup>63</sup> Therefore, heterogeneous mechanisms are more likely to be responsible, *e.g.*, on dust or trace impurities, or cavities. Table 4 shows a strong preference (>95%) for DH for both spontaneous nucleation and mechanical shock. In the present work, movement of the samples was unavoidable, and it's possible that what we call spontaneous nucleation was caused by mechanical shock. It is also possible that spontaneous nucleation is mediated by cavitation events caused by background particle radiation, such as cosmic rays.<sup>60</sup>

## 5. Conclusions

In summary, we have conducted a study of hydrate formation in aqueous sodium bromide. We found that both spontaneous nucleation and nucleation by mechanical shock strongly favoured sodium bromide DH crystals. Using sonocrystallization, the branching between AH and DH crystal forms could be rationalised in terms of the supersaturation with respect to each form, with reference to the solution phase diagram. Laser-trapping nucleation (in D<sub>2</sub>O) showed a complete preference for AH, and it was possible to nucleate solutions that were undersaturated with



**Fig. 7** Schematic diagram of the proposed mechanism for pulsed-laser NPLIN (see section 4.2). (a) A trace nanoparticle (NP) absorbs laser light and the heat transferred to the solution results in a superheated layer (SHL) of liquid. (b) Vaporization in the SHL results in a rapidly expanding vapor cavity. (c) The hot solvent is vaporized from an interfacial layer (IL) at the expanding gas–liquid interface. A crystal nucleus is formed in the IL, which favours AH due to the localised concentration and temperature.





respect to the AH solubility curve. NPLIN with single nanosecond laser pulses was almost entirely ineffective below the AH solubility curve, but gave a very strong preference for AH when supersaturated with respect to this phase. The results suggest that nucleation of crystal hydrates might in general be controlled by choice of method and knowledge of the phase diagram. From the results we have inferred new and useful details of the nucleation mechanisms involved. In particular, the results suggest that NPLIN causes cavitation events where the cavity interface is hotter, and therefore more dehydrating, than in sonocrystallization or nucleation by mechanical shock.

## Data availability

Data employed in this study are available *via* the Edinburgh DataShare repository (DOI: 10.7488/ds/3166x).

## Conflicts of interest

There are no conflicts to declare.

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