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Biomimetic hydrogels designed for cartilage tissue engineering

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Cartilage regeneration and repair remain a clinical challenge due to the limited capability of cartilage to self-regenerate. Worldwide, the costs associated with cartilage regeneration per patient are estimated on average £30 000 for producing and supplying cells. Regenerative approaches may include the use of cell therapies and tissue engineering by combining relevant cells, scaffolds and instructive biomolecules to stimulate or modulate cartilage repair. Hydrogels have been of great interest within these fields to be used as 3D substrates to cultivate and grow cartilage cells. Currently, biomimetic hydrogels with adequate biological and physicochemical properties, such as mechanical properties, capable of supporting load-bearing capability, are yet to succeed. In this review, biomaterials' advantages and disadvantages for the manufacturing of biomimetic hydrogels for cartilage regeneration are presented. Different studies on the formulation of cartilage-like hydrogels based on materials such as gelatin, chondroitin sulfate, hyaluronic acid and polyethylene glycol are summarised and contrasted in terms of their mechanical properties (e.g. elastic modulus) and ability to enhance cell function such as cell viability and GAG content. Current limitations and challenges of biomimetic hydrogels for cartilage regeneration are also presented.

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Introduction

Articular cartilage is a non-vascularised and low cellular density connective tissue in the joints. It is frequently damaged due to injury or normal wear and tear. Articular cartilage resists compressive forces and protects the underlying subchondral bone. It also provides a low friction surface for a diarthrodial joint to pivot. One of the most common joint diseases affecting articular cartilage is for example osteoarthritis, which affects approximately 300 million people worldwide and is the leading cause of disability among the elderly. This condition is caused by cartilage injury in the joints due to stress, especially in the weight-bearing joints. Due to its avascular nature, low cell proliferation, and migration, repairing cartilage has been one of the challenges for improving the clinical outcome of the patient.

Articular cartilage is made up of chondrocytes contained in the cartilage lacunae, residing in an extracellular matrix (ECM) produced by the chondrocytes. Chondrocytes reside in the cartilage in small cavities known as lacunae and reside cytoplasmically isolated from the neighbouring cells. Chondrocytes are normally rounded with a high matrix to cell volume ratio. In the cartilage, chondrocytes play roles in synthesizing and maintaining the ECM to prevent tissue deformation and supporting tissue function. They also influence epiphyseal plate growth by cell proliferation, matrix secretion, and cell volume increment during hypertrophy.⁴

Biomaterials Science Review

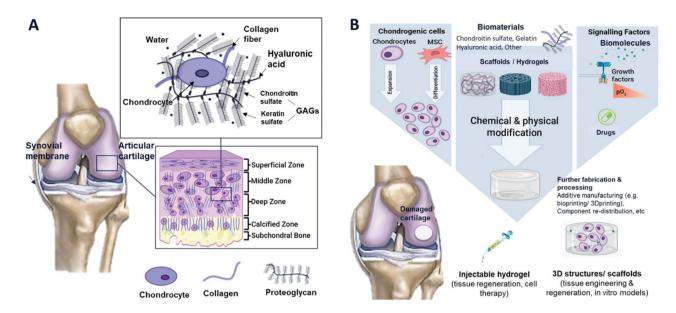


Fig. 1 Schematic illustrations of (A) cartilage composition and typical tissue zones and (B) the tissue engineering approach for cartilage repair.

The ECM is highly hydrated and contains proteoglycans (e.g. hyaluronic acid and glycosaminoglycans), collagen fibres and elastin. Glycosaminoglycans consist of chondroitin sulphate and keratin sulfate.⁵ ECM is essential for cartilage as it plays a role in cell signalling and it can regulate cell adhesion, growth, differentiation and migration. The ECM also maintains the physiological homeostasis of the cell microenvironment.⁶ Unlike most tissues, cartilage does not have blood vessels, nerves, or lymphatics. The composition of cartilage differs between four zones (Fig. 1A), which are mostly composed of proteoglycan content and collagen fibre alignment. In the superficial zone, collagen fibres are aligned parallel to the articulating surface, while in the middle zone, they are unaligned. In the deep zone, the fibres are radially arranged, and lastly, in the zone of calcified cartilage, an interface region between cartilage and bone is formed with the presence of mineralised fibres. The composition and organisation of each zone layer are to be considered when mimicking native cartilage tissues, as they impart distinctive zone mechanical properties, e.g. the compressive modulus increases with depth from the articular surface.

In superficial and mid-zone cartilage, chondrocytes function to produce collagen type II, IX, and XI, and proteoglycans of the ECM. Collagen type II makes up 90-95% of the collagen in the ECM and helps to stabilize the matrix, providing tensile and shear strength to the tissues.8 Collagen type IX and XI are less abundant in articular cartilage, and function to stabilize fibrillar collagen crosslinking. In the deep zone, chondrocytes are terminally differentiated and synthesize collagen type X, which plays a role in calcification.9 Type X collagen is also a hallmark for proteolytic enzyme production, which breaks down the cartilage ECM and allows for vascularization and calcification of tissues.8

Tissue engineering (TE) is an interdisciplinary field that combines materials science and cell biology to improve or

replace biological tissues. The main elements that build up TE are based on cell source, scaffold, and signalling molecules (Fig. 1B). The 3D porous scaffold acts as a template for tissue formation as it provides a suitable environment for the tissues or organs to grow. For example, scaffolds used in cartilage regeneration can be differentiated into sponges, membranes, and non-woven, non-injectable and injectable hydrogels. A hydrogel is a hydrophilic polymer that is capable of holding a large amount of water. 10 This property of hydrogels is crucial for mimicking the high water content (up to 80% 11) present in cartilage tissue. The scaffold is seeded with cells and optional signalling factors such as growth factors or external stimuli. 12 The use of scaffolds is also considered in other cartilage regenerative medicine approaches, such as in cell therapies like autologous or allogeneic chondrocyte implantation. The scaffold can be seeded with cartilage cells and placed in defective areas. So, the hydrogel is to be tuned to modulate cell behaviour such as cell migration and adhesion, providing temporary support for cellular function by polymer degradation.¹³

In cartilage tissue engineering, scaffolds or hydrogels with suitable properties to improve cellular function to regenerate cartilage and support load-bearing capability are yet to succeed. This review aims to give an overview of hydrogel usage for cartilage tissue regeneration, covering materials, techniques, and the future outlook of hydrogels as a tissue engineering approach.

The use of hydrogels in cartilage tissue engineering

Hydrogels are attractive scaffolds as they can be structurally similar to the ECM of various tissues, and can be delivered arthroscopically, making it a minimally invasive treatment. 13

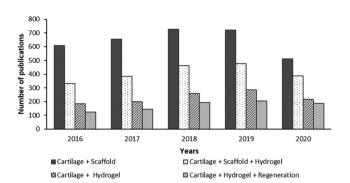


Fig. 2 The topic of research from the last five years in the "Web of Knowledge" on the use of hydrogel for cartilage tissue (last accessed on 25th September 2020).

Hydrogel properties can be tuned to a particular application including drug delivery and tissue repair. The usage of hydrogels for cartilage regeneration research has been increasing in the last decade. For instance, after a keyword search on the "Web of Knowledge" using a combination of "Scaffold", "Cartilage", "Hydrogel" and "Regeneration" on 25th of September 2020 (shown in Fig. 2), it was found that the number of research outputs on cartilage scaffolds on average nearly triples the number of studies on cartilage hydrogels. Based on this, the number of publications on hydrogels used as cartilage scaffolds is about 64% of the overall total of cartilage scaffold articles. Studies on hydrogels for the regeneration of cartilage have been steadily increasing up to 1.4 times between 2016 and 2020, representing about 74% of the total number of studies of cartilage hydrogel outputs.

Hydrogels offer the advantage of being tunable, so their properties can be modified to suit the application requirements. Some of the adjustable factors are the composition of the hydrogel, degree of crosslinking and methods of crosslinking that make the structure rigid, and density of the cells in the hydrogel. Ideally, for cartilage applications, the hydrogel composition should mimic the ECM whilst being able to withstand mechanical stress and loads, especially in load-bearing joints. ¹⁴ For instance, desired compressive stress between 0.4 and 2.0 MPa, tensile stress between 5 and 25 MPa, and shear stresses in the order of 3 MPa have been reported. ¹⁵⁻¹⁷

Biomimetic hydrogels for cartilage TE

Materials used to create biomimetic hydrogels

Broadly, hydrogels can be differentiated, according to the nature of the material composition, into natural, synthetic and biosynthetic. The natural hydrogels are commonly used for cartilage regeneration purposes; ⁵⁰ these include protein-based, polysaccharide-based, and decellularized hydrogels. The protein hydrogel includes collagen, elastin, fibrin, gelatin, and silk fibroin, while the polysaccharide hydrogel includes, for instance, glycosaminoglycans, alginate, chitosan, and agarose. ⁵¹ In addition, biodegradable synthetic materials have

been explored such as poly(ethylene glycol) (PEG), poly(N-isopropylacrylamide) (pNiPAAm), and poly(vinyl alcohol) (PVA). In addition, polyesters such as polylactide acid (PLA) and polycaprolactone (PCL) and its derivatives are incorporated into hydrogels to reinforce mechanical properties or as carriers for drug delivery.⁵² In Table 1, some of the common materials used for tissue engineering are summarised, reflecting their strengths, weaknesses and potential applications. In the last 5 years, innovative bioinspired hydrogels for cartilage applications have been formulated including mussel-inspired polydopamine-incorporated hydrogels^{32,53,54} to facilitate cell adhesion and tissue integration, incorporating sugar-based additives such as manuka honey to provide antibacterial properties⁵⁵ and improved mechanical properties,⁵⁶ and to mimick biological structures^{57,58} and networks⁵⁹ for improved mechanical and biological performance.

The use of crosslinking in biomimetic hydrogels

In hydrogels, the degree of crosslinking can control the swelling properties, while increasing the stability, and controlling the degradation rates and the level of porosity.⁶⁰ These are fundamental properties for tissue regeneration, since these will influence cellular function and the ability to provide support for new tissue formation. For instance, uncontrolled hydrogel swelling causes hydrostatic pressure in the surrounding tissue, which can cause inflammation and limited regeneration. The hydrogel polymer(s) chains can be crosslinked covalently or non-covalently (i.e. ionic bonds) involving either a chemical or physical reaction.⁶¹ Chemical crosslinkers are commonly used to establish covalent bonds between polymer chains, whereas physical crosslinkers facilitate physical interactions between chains or trigger chemical polymerizations. Cross-linking with free-radical polymerizations or copolymerization is commonly used by reacting hydrophilic monomers to multifunctional cross-linkers to produce cross-link junctions and physical interactions. The condensation reaction is often used to crosslink hydroxyl groups or amines with carboxylic acids present in proteins and some polysaccharides. For instance, the carbodiimide reaction may use the N,N-(3-dimethylaminopropyl)-N-ethylcarbodiimide (EDC) reagent to crosslink water-soluble polymers with amide bonds. The O-acylisourea intermediate, the carbodiimide activated acid, becomes deactivated fast, hence hindering the occurrence of amine reactions. The addition of N-hydroxysuccinimide (NHS) allows for more stable esters compared to the O-acylisourea intermediate and more reactivity towards amines.62

In situ crosslinking of injectable hydrogels allows for a minimally invasive procedure and cell encapsulation and transplantation, and enables the ability to match irregular defects. Different crosslinking methods can be used to prepare injectable hydrogels, including physical and chemical crosslinking such as photo-crosslinking, enzymatically crosslinked hydrogels, ion- or pH-sensitive hydrogels, click chemistry, etc. For instance, recent works on click-crosslinked modified HA hydrogels for cartilage tissue engineering have shown suitable mechanical properties and porous structures leading

Biomaterials Science Review

Table 1 Comparison of strengths and weaknesses of different biomaterials used in cartilage tissue engineering

| Natural material | | | | | |
|------------------------------------------------------------------------|-----------------------------------------------------------|-----------------------------------------------------------------------------------------------------------------------------------------|----------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|-----------------------|-------------------------------------------------------------------------------------------------------------------------------------------|
| Material | | Туре | Strengths | | Weaknesses |
| Collagen | | Protein-based | Low antigenicity, low inflamma excellent biological properties, biodegradability and biocompat | , | High cost, thrombogenic potential, low mechanical strength and modification difficulty ^{21,22} |
| Gelatin | | Protein-based | Low cost, minimal immunogen degradability and compatibility | city, | Stability decreases at high temperatures ²⁴ |
| Silk fibroin | | Protein-based | Excellent mechanical properties immunogenicity, ²⁵ and low thro | , low | Source difficulty and slow gelation ² |
| Glycosaminoglycans (hyaluronic acid (HA) : chondroitin sulphate | and | Polysaccharide- based | Mimics cartilage ECM composite biodegradable, ²⁸ reabsorbable, up, ²⁹ and binds signalling facto | ion, easy to scale | Rapid degradation <i>in vivo</i> and require crosslinking to be stable ^{31,3} |
| Alginate | | Polysaccharide- based | Non-toxic, non-inflammatory ar gelation ^{33,34} | | Poor cell adhesion and mechanical properties ³⁵ |
| Chitosan | | Polysaccharide- based | Anti-bacterial, low cost, biocomtuned biodegradability ³⁶ | patible and | Poor mechanical properties ³⁷ |
| Decellularized hydrog | gels | Protein-based | Recreate native ECM cartilage ³⁸ deliver biochemical signal | and can | Require complex processes and potential immunogenicity ³⁸ |
| Synthetic material | | | | | |
| Material | Strengths | | | Weaknesses | |
| Poly(ethylene glycol) (PEG) | Biodegradab | ole and non-immuno | genicity ³⁹ adjustable properties ⁴⁰ | Lack of adhes | sion support ⁴⁰ |
| Poly(glutamic acid) (PGA) Poly(lactic acid) and copolymers | Biodegradab mechanical Biodegradab solubility in | ole by hydrolysis, then properties, ⁴¹ porosity ole by hydrolysis, ⁴⁵ go organic solvents. ⁴⁷ | rmoplastic, adjustable variand localized inflammation and mechanical properties and localized inflammation and mechanical properties and localized indicate and reinforce hydrogels and localized in the model in th | can cause inf | ally crosslinked gel, hydrolysis produc Jammation and fast degradation ⁴⁴ -products can cause inflammation ⁴⁴ |

to high GAG expression and chondrogenic differentiation of human periodontal ligament stem cells (hPLSCs).64 Similarly, enzymatic crosslinking of hydrogels has drawn significant attention in the development of injectable hydrogels for tissue engineering, as it enables a quick gelation under physiological conditions with high specificity and low cytotoxicity.⁶⁵ Optimal enzyme selection still needs to be achieved to maintain cell bioactivity during encapsulation.⁶⁶ Some of the recent works on enzymatically crosslinked collagen-HA hydrogels showed promising results as injectable bone marrow mesenchymal stem cell (BMSC)-laden hydrogels for cartilage tissue regeneration.⁶⁷ These results were observed in vitro and in vivo when implanted in rats' cartilage defects, in which the hydrogel supported BMSC chondrogenic differentiation and hyaline cartilage repair. Similarly, studies on enzyme-crosslinked biomimetic hydrogels based on gelatin enriched with the cartilage extracellular matrix evidenced promising results in hyaline cartilage formation and GAG content in rabbit knee joint models.⁶⁸

Another approach is interpenetrating network (IPN) hydrogels, in which two or more polymers in the networks are partly interlaced on the molecular scale within the matrix; these are not covalently bonded and these polymers cannot be separated without breaking their chemical bonds in the matrix.69 Multiple networks such as IPN are better than a single network in terms of mechanical strength and swelling ability. 70,71 Broadly, the high water content tends to reduce the mechanical properties of single network hydrogels. Thus, novel approaches such as the "double network" (DN) have

been lately developed to improve their mechanical performance. This type of IPN hydrogel consists of a preparation first of a densely cross-linked polyelectrolyte network, followed by a neutral and loosely cross-linked (fully swollen) network. 72 This approach, initially promoted by Gong et al.,73 has led to promising results towards mimicking the native cartilage mechanical properties,74 including stiffness, high resistance to wear, and compression limit. 75,76 The polymer DN approach has been applied as a tissue-reinforcement mechanism by interpenetrating the damaged cartilage tissue, showing an increase of the tissue's equilibrium compressive modulus and wear resistance, while maintaining the cartilage volume when subject to harsh articulation conditions.⁷⁷ DN synthetic hydrogels have been used to mimic native mechanical properties and lubricity of cartilage obtaining exceptional cartilage-like properties suitable even for repairing chondral defects in the load-bearing regions of the body. 6 Lastly, DN hydrogels can be also enzymatically crosslinked for repairing cartilage. This was demonstrated recently by encapsulating human articular chondrocytes into biosynthetic DN hydrogels for the 3D bioprinting of cartilage engineering constructs, obtaining highly cytocompatible hydrogels with fast kinetics when compared to compositionally similar materials.⁷⁸

Mimicking cartilage composition

Cartilage is a highly hydrated tissue mostly composed of a combination of proteins and GAGs, such as collagen, HA, CS,

and KS. In particular, CS and KS (keratan sulfate) are two types of glycosaminoglycans (GAGs) that are normally attached to a protein core forming proteoglycans namely aggrecan.⁵ This KS/CS proteoglycan is linked to the HA core protein to form a larger proteoglycan aggregate. In particular, CS and HA are two types of glycosaminoglycans (GAGs) that are normally attached to a protein chain forming the proteoglycans found in cartilage.⁷⁹ Gelatin has been widely used in TE research as it is a derivate of collagen, a prominent ECM molecule of cartilage that has adhesion sites to chondrocytes.80 The combination of CS, HA, and gelatin displays great potential for cartilage TE as derivatives of the ECM. However, like most natural polymers, they lack good mechanical properties. To develop biomimetic hydrogels with improved mechanical properties, these are usually combined with water-soluble synthetic polymers, e.g. such as polyethylene glycol (PEG).81 Below, a further description is given of these materials used for cartilage regenerative approaches.

Chondroitin sulphate

Chondroitin sulphate (CS) is a polysaccharide molecule and the most abundant GAG found in the human body, which makes up 80% of GAGs in adult articular cartilage. 82,83 CS is a linear sulphated GAG with repeating p-glucuronic acid and p-N-acetylgalactosamine units.⁸⁴ It has the ability for tissue integration and exhibits an anti-inflammatory response.85 With ageing, the content of CS in cartilage decreases due to degeneration.86 CS is reported to be more effective in improving MSC chondrogenesis compared to heparan sulphates, especially in low stiffness. This is beneficial because a new matrix deposition accompanied by increasing mechanical properties occurs due to the neocartilage deposited by the cell.⁸⁷ Due to its high charge density, it has a high water content, which results in mechanical weakness. Hence, CS is usually combined and crosslinked with other materials for enhanced mechanical properties. 82,88,89 Chondrogenic regeneration has been widely studied using CS-based hydrogels. Previous studies on CS combined with pullulan hydrogels showed good viability of the encapsulated chondrocytes, displaying enhanced chondrogenesis in both self-crosslinked and enzymatically crosslinked hydrogels. 90,91 CS-based hydrogels have been shown to exhibit the highest gene expression and matrix accumulation throughout the cartilage zones compared to HAand type I collagen-based hydrogels.92

Hyaluronic acid

Hyaluronic acid (HA) is a non-sulphated GAG with a repeating disaccharide pattern, which is distributed within cartilage. In cartilage formation, HA is reported to induce MSC differentiation into chondrocytes, to maintain chondrocyte phenotype, and to increase the ECM deposition in cartilage. 116 However, the in vivo use of HA remains a challenge due to unsatisfactory cell adhesion ability as it can be degraded by hyaluronidase, nitrogen and reactive oxygen. It can induce inflammation due to foreign objects.93

Gelatin

Gelatin is a biodegradable natural polymer with singlestranded protein from collagen as a result of partial hydrolysis, preserving Arg-Gly-Asp (RGD) motifs for cells to adhere to. 94,95 It has been widely used in tissue engineering applications, as it is a low-cost polymer with minimal immunogenicity and tunable degradability.51 Furthermore, it is easily soluble in water at 37 °C and amphoteric. Gelatin consists of fractions derived from collagen alpha chains, maintaining a typical aminoacidic sequence of proline, hydroxyl proline and glycine.96 Promising results for in vitro MSC proliferation have been shown in the Pierce et al. study, where they use a gelatin-based hydrogel with few modifiable mechanical properties at various concentrations of ethyl lysine diisocyanate. Bone marrowderived MSCs were viable for nine days and displayed low toxicity.97 Hydrogels using bovine-sourced GEL-MA crosslinked with UV polymerisation displayed the most similar environment to native cartilage after 28 days of culture compared to porcine-derived GEL-MA. 98 Moreover, the addition of gelatin to other composite materials, such as alginate, has shown to increase their viability and chondrogenesis compared to alginate alone.99

The use of synthetic and water-soluble polymers such as PEG has become attractive due to their biocompatibility and ability to prevent the nonspecific absorption of protein and neutral charge. 100 They can act as spacer agents, hence resulting in a more effective and controllable way to link two molecules.81 PEG is the most used synthetic hydrogel for tissue engineering, given its inert nature chemically and physically.101 It is also biodegradable and non-immunogenic. PEG also displays a non-toxic and highly soluble nature with good mechanical properties. 102,103

Biomimetic hydrogels: mechanical and biological properties

Biomimetic hydrogels might mimic cartilage ECM composition and/or organisation to achieve biological and mechanical properties similar to those of the native tissue. For instance, natural materials such as CS and gelatin are nontoxic with tunable biodegradability; however, they have poor mechanical properties, which limit their use for tissue engineering. 104 For this reason, the combination of natural materials with synthetic materials is used for achieving improved overall hydrogels' biological and mechanical properties. The reported compressive modulus of articular cartilage¹⁰⁵ is 0.02-1.16 MPa in the superficial zone and 6.44-7.75 MPa in the deep zone, while in osteoarthritic cartilage, it may range from 2 MPa to 30 MPa.¹⁷ A summary of selected works, in which natural materials such as CS, gelatin and HA are combined with synthetic ones such as PEG, is shown in Table 2 and extended to Fig. 3.

Data related to mechanical (i.e. compressive modulus) and biological properties (i.e. cell viability and GAG/DNA content)

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 Table 2
 Summary of selected hydrogels based on PEG, CS, HA and gelatin for cartilage regeneration

| Hydrogel | Crosslinkers | Time-point | Cells | Findings | Viability, ECM Production ($\mu g \mu g^{-1}$), compressive modulus |
|--------------------------------------------------------------|--------------------------------------------------------------------------------|----------------------------------------------------|--------------------------------------------------|------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|
| CS-based (with graphene) ¹⁰⁶ | EDC/NHS | 3 weeks | MSCs | EDAG-CS porous hydrogels exhibit higher mechanical properties, but degrade slower than CS-CS. EDAG-CS and CS-CS produce type II collagen and MSCs do not differentiate to endochondral ossification | Cell viability and ECM production not reported Compressive modulus: 1.12 kPa |
| Carboxymethylated pullulan CMP-TA/ CS-TA ⁹⁰ | Enzymatic crosslinking (horseradish peroxides and ${\rm H}_2{\rm O}_2$) | 14 days | Porcine auricular chondrocytes | Hydrogel with a CMP-TA/CS-TA weight ratio of 3/1 provided best host tissue-mimetic microenvironment. | Viability: 80–95% (CMP-TA), >95% (CMP-TA/CS-TA) (CMP-TA/CS-TA) ECM production not reported Compressive modulus (CMP-TA): 0.96 kPa |
| $ m GEL$ -base $ m d^{107}$ | UV photo-polymerization | 3 days, 4 weeks | Chondrocytes | Supplementation of HA to increase viscosity and PCL polymer for structural support enables the build-up of a layered hydrogel structure. Cell viability detected until 28 days using GAG/DNA. | Viability: 73 ± 2% (GEL-MA), 82 ± 8% (GEL-MA-HA) (day 3) GAG/DNA: 7.1 ± 3.3 (GEL-MA), 10.5 ± 5.2 (GEL-MA-HA) (week 4) Compressive modulus: 32.28–53.54 kPa (GEL-MA) |
| $	ext{CS-PEG}^{108}$ | UV photo-polymerization | 24 hours, 3 weeks | Juvenile, adult, and osteoarthritic chondrocytes | 7% w/v of PEGDA and 3% w/v. Deposition of ECM and long-term cell viability are reported. | Viability: 94% ECM production not reported Compressive modulus: 0.73-1.04 kPa (day 1, week 3) |
| $\mathrm{CS	ext{-}PEG^{109}}$ | Photo-polymerization | 3 weeks | Neonatal bovine chondrocytes | Dual gradient hydrogel with interconnected chambers (20% w/v PEG, 3% w/v CS; zone 1; to 2% w/v PEG, 8% w/v CS; zone 5) with better mechanical properties, cell proliferation and matrix deposition than CS hydrogel caradient alone. | Viability: >90% in all zones (7 days) GAG/DNA: 40.58–53.72 kPa (zone 1– zone 5) Compressive modulus: 9.2 kPa–124.7 kPa (zone 1–zone 5) |
| CS-PEG PEG-CS-MMP-pep PEG-HA ¹¹⁰ | UV photo-polymerization | 2, 4, 6 weeks | D1 mouse bone marrow progenitor cell | Higher collagen II content was detected in the superficial layer (PEG:CS:MMP-pep), collagen X content and GAG production were increasing from the superficial layer, transitional (PEG:CS) to the deep layer (PEG:HA) mimicking the cartilage layer. | Viability: 86% (PEG-CS-MMP-pep), >95% (CS-PEG), 84% (PEG-HA) GAG/DNA: PEG-CS-MMP-pep: 2.6; CS-PEG: 2.8; PEG-HA: 6.6 (2 weeks) Compressive modulus: PEG-CS-MMP-pep: 199.71 (120–209 kPa), CS-PEG: 203.95 (121–270 kPa); PEG-HA: 942.83 (271–1227 kPa) |
| CS-PEG ¹¹¹ | EDC/NHS | 24 hours (in vitro) 4 weeks (in vivo) | Bovine articular chondrocytes | Cells within hydrogel retain viability and metabolically active, minimal inflammatory response compared to PEG, stronger adhesion compared with fibrin <i>in vivo</i> . | Viability: 93–96% ECM production not reported Compressive modulus: 11.31–22.28 kPa |
| CS-PEG ¹¹² | UV photo-polymerization | 24 hours, 3, 6 weeks | MSCs | PEG-CS gels have 10 times stronger adhesion to cartilage tissue compared to fibrin, and decreased inflammatory response compared with PEG. | Viability: 79% (PEG), 84% (CS-PEG) (24 hours) GAGS/DNA: PEG: 10.75 (week 3), 20.12 (week 6). CS-PEG: 18.11 (week 3), 36.18 (week 6) Compressive modulus not reported |
| PEG-LA-DM PEG-DM ¹¹³ | UV photo-polymerization | 6 weeks (in vitro) 6, 12, 18 weeks (in vivo) | Juvenile swine articular chondrocytes | Encapsulated chondrocytes in 60/40 and 70/30 PEG-LA-DM/PEG-DM polymer ratios Improved cartilage matrix formation of aggrecan and collagen type II/VI | Cell viability and ECM production not reported Compressive modulus: 60–670 kPa (ref. 92) |
| PEG-HA ¹¹⁴ | Enzymatic crosslinking (HRP and $\rm H_2O_2$) and Diels–Alder click chemistry | 7 days | Chondrogenic teratocarcinoma cells | Enzymatic crosslinking hydrogels result in 5 minutes of gelation time, has great shape memory and antifatigue properties | Viability: >95% No ECM production reported Compressive modulus: 60.7–109.4 kPa |

Table 2 (Contd.)

| (100) | | | | | |
|----------------------------------------|-----------------------------------------------------------------------------------------------------------------|-------------------------|-------------------------------------------------------------------------------------------------------------------|--------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|--------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|
| Hydrogel | Crosslinkers | Time-point | Cells | Findings | Viability, ECM Production ($\mu g \ \mu g^{-1}$), compressive modulus |
| PEG-GelMa | Photo-polymerization | 7 days, 3 weeks | Human MSCs | Inkjet bioprinted MSCs of PEG-MA show better mechanical properties than PEG, increased 63% under chondrogenic differentiation after 21 days. | Viability: 85.87% (PEG-GelMa), 82.43% (PEG) GAGS/DNA: 73.81 (PEG-GelMa), 67.64 |
| PEG^{115} | | | | | Compressive modulus: 36.70 kPa |
| $\mathrm{GEL-CS}^{116}$ | EDC/NHS and click chemistry | 1, 3 and 7 days | Fibroblasts and macrophages | Overall good cytocompatibility, mimicking cartilage ECM composition | (PEG-Gelma), 43.31 KPa (PEG) Viability: 10% CS-gelatin (79% MTT); Compressive modulus: not measured. |
| GEL-CS ¹¹⁷ | Chemical crosslinking (glutaraldehyde, dithiothreitol/ 2-propanol); and photo- crosslinking (UV-light) | 21 days | Human mesenchymal stem cells (MSCs) | Spatially patterned µRB scaffold enabled MSCs to produce cartilage with biomimetic zonal biochemical and mechanical properties | ECM production: not measured Viability: not measured quantitatively. Compressive modulus: trilayer µRB up to 456 kPa. ECM production: sGAG per d.w trilayer µRB up to 0.08 µg µg ⁻¹ |
| GEL-HA ¹¹⁸ | Photo-crosslinking (UV-light) | Up to 8 weeks | Human bone marrow-derived mesenchymal stem cells (hBMSCs) and <i>in vivo</i> (New Zealand white rabbits) | MSCs chondrogenesis, scaffolds at 9:1 (% w/v) ratio mGL/mHA lowest hBMSC hypertrophy and highest glycosaminoglycan production, supported cartilage formation <i>in vivo</i> after 12 weeks | (1.5-7.5% dry weight) Viability: not measured quantitatively. Compressive modulus: up to 25 kPa (8.5/1.5 ratio mGL/mHA) ECM production: up to 17 µg GAG/ |
| GEL-HA ¹¹⁹ | Photo-crosslinking (UV-light) | 1, 4, 7 days | Chondrocytes and <i>in vivo</i> (nude mice) | At a relatively low cell-seeding concentration, homogeneous tubular cartilage regenerated with a tracheal shape, mechanical strength and cartilage ECM deposition | construct Cell viabout 98%. Storage modulus $(G') = 1$ kPa. ECM production: no reported <i>in vitro</i> but <i>in vivo</i> (about 70 mg 2 1 LDTM |
| GEL-HA ¹²⁰ | Enzymatic crosslinking (HRP and ${ m H}_2{ m O}_2)$ | Up to 14 days | Mesenchymal stem cells (MSCs) | 70% HA and 30% gel most promising for chondrogenesis, best mechanical properties | scanold and 200 mg g 101 Exp) Viability not measured quantitatively, cell proliferation decreases in hydrogels rich in HA. Compressive modulus (G') up to 789 ± 220 Pa. ECM production = GAGs about 80% |
| Collagen-CS-HA (CCH) ¹²¹ | Chemical crosslinking (Genipin, 0.75 mM) | 4 weeks | Allogeneic chondrocytes (in vivo Rabbits model) | CCH hydrogel provides structural and mechanical similarity to cartilage; used as a carrier of chondrocytes; helps cell growth and matrix synthesis | In the 301/0 mxture and HA get. Viability not measured quantitatively, cell proliferation increases in CCH hydrogels. Compressive modulus increases (CCH about 51.53 ± 0.54 kPa (ref. |
| HA-GEL-CS ¹²³ | Diels-Alder click chemistry | N/A | N/A | HA-furan GEL-furan, MAL-PEG-MAL | GAG/DNA ratio about 4 (7 days, approx. 6 µg µl ⁻¹) Cell viability and ECM production |
| HA-GEL-CS ¹²⁴ | Click chemistry (Cu(i) as the catalyst) | Up to 7 days | Chondrocytes | HA-GEL-CS with EDC NHS Good reactivity under very mild conditions by click-chemistry. <i>In vitro</i> cell culture confirmed that chondrocytes could adhere and proliferate on the | For the policy of the policy o |
| GEL-MA GEL-HA | Photo-crosslinking (UV light) | 1 day, 5 and 8 weeks | Human chondrocytes | nydrogel. The addition of CS and HA to GEL-MA can enhance the differentiation, improved mechanical properties, and ECM distribution. Significant COL2A1 expression | ECM production: not measured Viability: 80–90% GGE-MA), 140 (GEL-HA), 92.7 (GEL-CS), 125.3 |
| GEL-CS GEL-HA-CS ¹²⁵ | | | | | (UEN-TA-C.5) (WEEK 8) Compressive modulus: 20.17 kPa (GEL-MA), 31.08 kPa (GEL-HA), 21.66 kPa (GEL-CS), 32.57 kPa (GEL-HA-CS) |

CS: Chondroitin sulfate; CMP: Carboxymethyl pullulan; DM: Dimethacrylate; EDAG: Ethylenediamine graphene; GEL: Gelatin; GELMA: Gelatin methacrylates; HA: Hyaluronic acid; LA: 4, 5 lactic acid; MA: Methacrylates; MMPpep: Matrix metalloproteinase-sensitive peptides; PEG: Polyethylene glycol; TA: Tyramine.

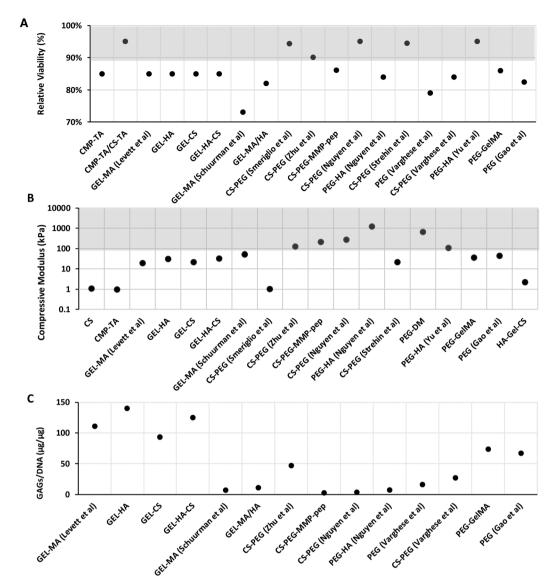


Fig. 3 Compressive modulus, viability, and GAG/DNA of selected hydrogels. (A) Cell viability; (B) compressive modulus; and (C) GAG/DNA of hydrogels for cartilage regeneration; gray-shaded areas indicate the desired value for cartilage applications.

of biomimetic hydrogel studies were collected from different journal databases (PubMed, Web of Science, and Google Scholar). From this review, it was found that most of the hydrogels found in the literature have a compression modulus ranging from 10 to 1000 kPa, meaning that formulated hydrogels are below cartilage values by at least a factor of 10 (Fig. 3A). Particularly, it can be observed that cell-free hydrogels based on PEG-HA¹¹⁰ (217-1227 kPa) show the highest compressive modulus when compared to PEG-DM¹¹³ (60-670 kPa) and CS-PEG¹¹⁰ (121-270 kPa). Interestingly, materials based on CS-PEG can show a very low compressive modulus (e.g. 0.73 kPa);¹⁰⁸ this can be improved by varying the crosslinking method. 109-111 Increased PEG concentration will result in a stiffer hydrogel.

The hydrogel should provide to the cells a suitable microenvironment that stimulates ECM production and new tissue formation. For instance, scaffolds that are too stiff can affect biological behaviour, such as changing the phenotype of the cells and driving MSC differentiation into bone cells. 126,127 Moreover, it was found that by increasing substrate stiffness, chondrocyte ECM production is down-regulated while promoting adhesion and stress fibre formation. 128,129 However, if the modulus is not adequate in early time points, chondrocyte will produce ECM that can strengthen the hydrogel structure over time and withstand higher compressive forces. 130,131

Data on cellular viability (using LIVE/DEAD assays) collected from 24 hours up to 7 days following cell seeding are shown in Fig. 3B. In general, hydrogels show good cell viability (higher than 70%), particularly those based on proteoglycans like CS and HA. From relative viability data, the overall high viability was shown for hydrogels based on CMP-TA/CS-TA,90 CS-PEG, 108,110,111 and PEG-HA, 123 evidencing higher survival

Review
(>90%) when compared to hydrogels such as GEL-MA¹⁰⁷ and absence of synthetic

PEG¹¹² (<80%). Particularly, based on the latter, GEL-MA showed the lowest cell viability amongst the group, in contrast to other gelatin-based hydrogels. 132 This effect may be due to the methacrylate process, which causes cytotoxicity. Methacrylate toxicity has been reported in several in vitro studies where glutathione (GSH), a major antioxidant that prevents oxidative stress, was removed. Increased oxidative stress caused by adduct formation of methacrylate results in cell toxicity. 133,134 Regarding synthetic PEG-based hydrogels, these do not contain cell adhesion motifs and so increasing the PEG concentration or polymer block length in hydrogels might result in decreased or impaired cell adhesion and protein adsorption. 135,136 GAG/DNA content is presented as the total accumulated GAG content normalised to the total DNA content of a selection of biomimetic hydrogels in Fig. 3C. In terms of ECM production, the gelatin-based hydrogel seems to stimulate GAG secretion, except for GEL-MA. 107 The gelatinbased hydrogel showed the highest levels of GAG/DNA deposition; particularly, GEL-HA had the highest GAG accumulation, followed by GEL-HA-CS. 125 The use of HA has been shown to enhance ECM remodelling into a more ordered collagen deposition, and increased the mRNA expression of cartilage-specific genes. 137,138 Consistent results were shown where the addition of HA produces a higher GAG content. 107,110,125 However, it appears to be concentration-dependent, where the chondrogenic properties are only shown at low concentrations, while no effect or a negative effect on chondrogenesis is observed at higher HA concentrations. 107

In contrast, the lowest GAG/DNA production was found when the cells were incubated in hydrogels containing PEG. 110-112 One way to overcome this problem is to incorporate other polymers. Varghese et al. have shown that incorporation of CS into PEG hydrogels results in enhanced chondrogenic gene expression and cartilage matrix production compared to PEG alone. 112 Moreover, Zhu et al. hypothesised that a stiffer hydrogel could be made by increasing the PEG and CS concentrations in a layer-by-layer manner, resulting in more ECM production using a mechanical gradient hydrogel via cellular remodelling of the ECM by degradation. 109 Interestingly, the combination of GEL-MA and PEG favours ECM production and promotes MSC chondrogenic differentiation. 115 Bryant et al. reported that incorporation of degradable crosslinkers can promote GAGs to diffuse out of the gel without sacrificing the mechanical properties. 139 In contrast, studies have suggested that CS can inhibit ECM production, despite it not impacting the chondrocyte viability. 109,112 This phenomenon may be caused by the negative charge of CS that attracts free cations from the culture medium, which increases hydrogel osmolarity and impacts cell growth negatively. 140 Moreover, the addition of signalling molecules such as TGF-β1 can be considered for future research to improve cellular function such as cell proliferation and ECM secretion in syntheticbased hydrogels. 141

In the last few years, improved cartilage ECM-like hydrogels based on CS, HA and gelatin have been formulated in the

absence of synthetic polymers such as PEG. Improved mechanical and biological performance has been achieved by exploiting the recent advances in physical and chemical crosslinking including click chemistries, photo-crosslinking, enzymemediated crosslinking, and interpenetrating double networks. Mechanical properties similar to native cartilage have been obtained for biomimetic hydrogels based on ECM-derived proteins, 121,125 achieving a compressive Young's modulus up to 456 kPa for spatially patterned µRB scaffolds based on CS-GEL hydrogels. 117 In this context, excellent mechanical and biological performance including GAG deposition and cellular viability have also been obtained for hydrogels that combine ECM-derived materials such as HA and GEL. 119,121,125 These properties were achieved by combining the biological cues present in the native cartilage ECM while reinforcing their structure with the presence of non-toxic crosslinking strategies without compromising hydration or cyto-/biocompatibility. In general, naturally derived biomaterials and combined ECM proteins in hydrogels (e.g. CS, HA, gelatin) are known as ideal biomimetic sources for cartilage tissue engineering. However, some of the key limitations, including weak mechanical properties for mimicking the surrounding tissues and control of structure and degradation rates, have limited their applications. In this overview, we have shown that recent advances in chemical or physical modifications and fabrication methods are capable of overcoming some of these limitations, demonstrating that these materials can mimic native ECM tissues and support cartilage regeneration while showing superior functional performance when compared to single or biosynthetic formulations.

Future perspective on cartilage tissue engineering

For the past 5 years, there has been growing interest in hydrogel research for cartilage regeneration. Moreover, it has been shown that hydrogels can provide a suitable environment for chondrocytes and MSCs to grow from *in vitro* and *in vivo* research. Hydrogels capable of mimicking properties and biological activities, structure and organisation of the native tissue are needed for developing TE strategies, disease models and cell therapies.

In vitro cartilage tissue models

Zonally stratified structures similar to native cartilage can be achieved by exploiting combinations of materials and composite hydrogels processed additively. Therefore, a gradient hydrogel processing technique, such as bioprinting and controlled fluid mixing, allows the creation of multiple layers with a physical and chemical composition that mimics cartilage zones. Biomimetic multiphasic hydrogels and structures have been developed to mimic the native osteochondral tissue properties, including monophasic, bi-phasic and tri-phasic structures. For instance, mussel-inspired hydrogels with a bilayer structure have shown potential to repair osteochondral

Biomaterials Science Review

defects following implantation in rabbit knee joints. 145 Another example is the biphasic CAN-PAC hydrogel for the regeneration of osteochondral defects prepared by thermally initiated free radical polymerization, which showed a low inflammatory response and ingrowth of newly formed tissue when subcutaneously implanted in rats. 146 Moreover, advances in tissue engineering by combining hydrogels and bioprinting technologies to manufacture relevant cellular microniches present new exciting avenues for tissue modelling, ageing and disease mechanism discovery and therapy towards reducing the need for costly and unethical animal models. Photo-crosslinkable GEL-HA has been fabricated as a porous scaffold for cartilage repair with a suitable internal pore structure and high mechanical strength via photocuring 3D printing and lyophilization.¹⁴⁷ The encapsulation of chondrocytes into the scaffolds promoted mature cartilage regeneration with a typical lacunae structure and cartilage-specific ECM both in vitro and in vivo. Recent works on the bioprinting of biomimetic hydrogels for cartilage applications showed a great interest in using functionalised natural components 148 such as gelatin, chondroitin sulphate, and hyaluronic acid towards recapitulating the complex zonal microarchitecture of native hyaline cartilage, 129 and creating anatomically accurate constructs of stratified cartilage tissues and osteochondral tissue interfaces. 149,150

Cartilage is a mechanoresponsive tissue, and dynamic stresses play an important role in cartilage development and homeostasis.¹⁵¹ However, the mechanotransduction pathways, including mechanisms and parameters involved are poorly understood, presenting a major growth area with the potential for discovering new therapeutic approaches in cartilage ageing and disease. Still, advancements for tissue engineering require more reliable assays for the analysis and characterisation of cell behaviour in 3D structures, and for assessing the influence of biomechanical cues on cartilage development. In terms of assessing 3D cartilage-like structures, there is a fundamental need for more reliable assays for *in vitro* assessment and standardisation of protocols, as most available products are specifically made for 2D culture.

In silico modelling of cartilage tissue

To achieve the desired properties of the hydrogels, generally a significant amount of iterations in multi-variant systems is needed for the design, manufacturing and characterisation of these systems. One promising strategy is the emerging concept of rational design that is based on computationally guided biomaterial design strategies. These have been shown to be very suitable when multiple design variables need to be considered. For selecting a biomaterial with suitable mechanical properties, Bas *et al.* developed a numerical model-based approach for the rational design of biomimetic soft network composites to be used in cartilage regeneration. They established a design library capable of predicting the compressive modulus of the designed biomaterials using numerical tools from compressive modulus and Poisson's ratio data of hydrogels found in the literature. This library was

created specifically for soft hydrogels ($E = \sim 10-50$ kPa) that can be applied for encapsulating different cells such as human chondrocytes.

Furthermore, a computational model can help to investigate and clarify some of the most significant mechanisms that influence the growth and degradation of the hydrogels used in regenerative medicine. In this regard, an interesting mechanics-based model of tissue growth in degradable cell-laden hydrogels has been reported recently by Sridhar et al. 155 In this continuum-based multiphasic model, the authors assumed that, after encapsulating cells in a PEG-based hydrogel, hydrolysis of polymer crosslinks was taking place following a first order kinetic law. 156 Then, consequently to the crosslink cleavage, they assumed a release of ECM precursors from the cells to assemble large ECM molecules of aggrecan and collagens. Interestingly, applying their model, they found that well-linked and dense cell agglomerates combined with large areas of weak cross-linking around cells could generate the right conditions/ factors for the formation and growth of new tissue while maintaining structural integrity. Similarly, Akalp et al. 157 developed a model for interstitial growth, based on mixture theory, which could be used to investigate the influence of cell-laden hydrogel degradation. This was also due to the crosslink cleavage mediated by sensitive enzymes and the transport/diffusion of ECM molecules released by the cells. Upon application to the cartilage tissue, their numerical results showed that the enzymatic degradation produced the presence of a very localized degradation front spreading away from the cells, which could produce immediately a front of growing neo-tissue. Even if these studies utilised real experimental data for the model validation, they did not take in to account the in vivo conditions. In this scenario, from a more etiopathological view, Stender et al. 158 developed a FE model to study the changes of osteoarthritisinduced articular cartilage damage and the remodelling of subchondral cortical and trabecular bones.

Finally, in spite of the efficiency of the computational models in identifying potential growth mechanisms and supporting the experimental plan, the future models are expected to be able to be used for personalised tissue engineering. This is since the changes in cell behaviour can strongly influence the formation and growth of neo-tissues as well as the dynamic degradation mechanisms of the hydrogels. ¹⁵⁹

Cell therapies for cartilage regeneration

Degenerative skeletal diseases such as osteoarthritis affect more than 200 million people worldwide. Current strategies of joint replacements lead to the removal of cartilage and healthy bone tissues. Moreover, joint devices and implants have a limited lifespan, requiring revision surgeries within 20 years, so there is a need to move from tissue replacement to tissue regeneration. Matrix autologous chondrocyte implantation (MACI) cell therapies are as expensive as £17 740 (ChondroCelect®, withdrawn from the European Union) per procedure with insufficient evidence of cost-effectiveness. 160 One of the major limitations of MACI is increased timings for patient rehabilitation, caused by poor mechanical and struc-

tural integrity of implanted materials before tissue regeneration. ¹⁶¹ MACI, which was approved in Europe, is also suspended as of 2017.

Autologous chondrocyte implantation (ACI) is available in Europe and the UK (since 2017) for treating symptomatic articular cartilage defects (>2 cm²) of the knee. However, ACI is successful for approximately 80% of patients, but for the remaining 20% can produce tissue delamination and hypertrophy of the implantation site. 163 Similarly, a scaffoldfree spheroid, Chondrosphere (Spherox), has been approved by the EMA in 2017. Chondrosphere versus MACI was £18 000 per quality-adjusted life-year gained. 164 For these cell therapies, new biomimetic hydrogels and materials have the potential to improve the cell therapy outcomes, while decreasing the associated costs. Advanced biomaterials are needed to provide the best environment and support for cartilage cells to function. It is important to note that products containing active biomolecules and/or genes require more complex procedures, regulation, and funding which can limit the sustainability.¹³

In the US market, FDA-approved cell-free hydrogels are Gel-One®, Synvisc and Synvisc-One®. 165,166 In contrast, products that are hydrogel-based with autologous chondrocytes include CaReS®, a type I collagen, and Cartipatch®, which uses an agarose–alginate hydrogel. Cartipatch® has a phase III study terminated and CaRes® is only available in a few selected European countries, Iran, China, and Turkey 167,168

In summary, this study investigated the mechanical and biological properties of widely used hydrogel materials such as PEG, CS, gelatin, and other materials, for cartilage regeneration. From the literature review, we find that the overall viability of hydrogels based on PEG, CS, or gelatin is high. GAG-based hydrogels namely CS and HA have the highest viability compared to the rest. In general, PEG-based hydrogels have the highest Young's modulus though it is still not adequate compared to the natural articular cartilage compression modulus. Incorporation of additional materials may influence the mechanical properties and viability of chondrocytes. Variation of stiffness and material composition affect chondrocytes to form the matrix. However, research on matrix stiffness and chemistry of hydrogels on cell behaviour is still conflicting. Further research on the effect of stiffness on cellular proliferation, differentiation, ECM production, and migration is needed. Therefore, future studies should focus on investigating the physical, chemical, and biological properties of PEG, CS, and gelatin for use in hydrogels as a scaffold for cartilage regeneration.

Conflicts of interest

The authors declare no conflicts of interest.

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