



Cite this: *RSC Adv.*, 2019, 9, 12585

Construction of anisotropic fluorescent nanofibers assisted by electro-spinning and its optical sensing applications†

Fa-Heng Zhang,^a Rui-Xue Jiang,^c Wei Cao,^b Bin Du,^{id} Ding-Yuan Cao,^d Zhi-Jun Ding^{id}*^{ab} and Zhi-Jun Li*^a

Fixing the gap between “nano-scaled” pieces and “product-scale” materials, devices or machines is an ineluctable challenge that people have to tackle. Herein, we show that combining self-assembly and electrospinning processes results in the fabrication of anisotropic fluorescent nanofibers (PDI@PVDF) in which the well-defined rod-like perylene bisimide derivative assemblies are embedded in a highly oriented way along the axis of the poly(vinylidene fluoride) (PVDF) fiber. Compared to fragile individual PDI assemblies, the electrospinning anisotropic fluorescent PDI@PVDF nanofibers not only maintain high sensitivity for aniline vapour but also exhibit an unexpected short response time for both quenching and recovering. The results demonstrate that electrospinning assistance is a versatile and effective strategy to maintain the anisotropy of fluorescent nanomaterials, building a bridge between self-assembled nanorods and practical materials.

Received 21st January 2019

Accepted 3rd April 2019

DOI: 10.1039/c9ra00502a

rsc.li/rsc-advances

1. Introduction

For decades, scientists have paid more and more attention to developing technologies and processes to assemble nanometer-scale pieces into systems, components, or materials.¹ As a basic structure among these micro/nanostructures, 1-dimensional (1D) nanostructures such as nanorods,^{2a} nanotubes,³ nanowires⁴ and nanostrands^{2b} exhibit unique and attractive “nano-scale” behaviors⁵ including quantized current–voltage behavior,⁶ dramatically lower melting points and significantly higher specific heats, highly anisotropic geometry, large surface-to-volume ratio, and carrier and photon confinement. However, they tend to lose these potentially beneficial traits when they are manufactured at larger “product-scale” dimensions, failing integration into devices and machines. This is mainly because the loss is in terms of the structural integrity and geometric uniformity.^{5a} The bottom-up approach, through self-assembly and supramolecular chemistry, provides an exciting alternative route for the fabrication of various artificial materials with special micro/nanostructures in a hierarchical way.² Up to now, solution-based self-assembly is indeed the

main strategy for producing 1D structures.^{2b} The main challenge^{4a} is that the thermodynamic assembly processes⁷ usually take quite a long time (measured in hours even days) to achieve equilibrium.

On the other hand, a number of techniques, including, self-assembling,^{2a,2b,2d} nanolithography,⁸ electrospinning^{2c,9} and physical drawing,¹⁰ have currently been developed for the fabrication of 1D structure⁴ at “product-scale”. Especially, the electrospinning technique can boost the fabricating of polymer fibers with micro- or nanometer diameters within milliseconds from a broad range of polymer materials.^{2c} The millisecond time-scale could prevent excessive growth of self assemblies by shortening (disrupt) the assembly process meanwhile the micro/nano-scaled diameter provides axial confining enclosure (interspace). For this reason, electrospinning shows potential applications in rapid self assembly of building blocks into high-ordered nanostructure.¹¹ However, in most electrospun nanofibers, the dopants dispersed randomly in the polymer nanofibers without obvious ordered structures, wherein the functional dopants are not located in an ordered manner that like in their individual self assemblies.¹² This is mainly because that, due to the loss in terms of the structural integrity and geometric uniformity, electrospinning process usually lead to some degree of function degradation that we have demonstrated above. In this context, how to maintain the anisotropy of dopants in electrospinning assistant 1D nanomaterials is another challenge (Scheme 1).

Based on their high thermal stability and photostability, perylene bisimide derivatives have been proven as excellent supramolecular building blocks and widely used as liquid

^aResearch Institute of Chemical Defense, Beijing 102205, China. E-mail: zhijunding01@gmail.com

^bState Key Laboratory of NBC Protection for Civilian, Beijing 102205, China

^cCollege of Chemical Engineering, China University of Petroleum Huadong, Qingdao Campus, Qingdao 266580, China

^dNankai University, Tianjin 300071, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9ra00502a





Scheme 1 Schematic representation for the hierarchical assembly in solid-state and electro-spinning assistant confined self-assembly of *N,N'*-bis(1-hexylheptyl)perylene-3,4,9,10-tetracarboxylbisimide (PDI) for optical sensing applications.

crystals,^{11,14} molecular wires,¹⁵ semiconductor,¹⁶ and transistors.¹⁷ Optical sensing behaviors¹⁸ of perylene bisimide nanoarchitectures in both solution and solid-state had been detailed reviewed by Zang and coworkers.^{2b} Herein we report a strategy that combine self-assembly and electrospinning processes to construct a type of **PDI@PVDF** nanofibers. Structurally, the bricks (the organic assembly of **PDI**) are arranged in an ordered manner. Functionally, the response time of the nanofiber towards aniline was boosted from minutes to seconds.

2. Results and discussion

2.1 Self-assembly of PDI

We investigated the concentration dependent hierarchical self-assembly behavior of individual **PDI** in solution. The UV/vis absorption and fluorescence emission spectra of **PDI** were recorded in chloroform at different concentrations. The absorption spectrum of **PDI** at lower concentration ranges from 0 to 1.07×10^{-5} M showed sharp absorption bands at 458, 491, and 526 nm (Fig. S2a, ESI[†]), indicating that the **PDI** molecules remain monomeric phase at the relative lower concentration ranges. Higher concentration led dramatically rise of UV/vis absorption spectra and finally resulted a saturated spectrum.

The fluorescence spectrum has a mirror-image relationship to the absorption spectrum¹⁹ and provides more information of the aggregate (Fig. S3, ESI[†]). At the lower concentration ranges the intensity of the emission increased upon gradual augmentation of the concentration, while the intensity of the emission decreased due to π - π electronic coupling as free molecules transforming into self-assemblies at the higher concentration ranges¹³ (Fig. S4, ESI[†]). The normalized emission spectrum showed a continuous red shift indicating the formation of the **PDI** aggregation underwent an assembly process (Fig. S2b, ESI[†]). Another evidence of the self-assembly process was the concentration dependent color change. The emission color changed gradually from bright green (for the totally free dispersed molecules) to bright red (for the self-assemblies) (Fig. S4 inset, ESI[†]).

This self-assembly process was further investigated in solid-state. As demonstrated by scanning electron microscopy (SEM) image in Fig. 1, when the solvent was evaporated in vacuum to give a red large agglomerates in which the sub-micrometer aggregates were randomly embedded (Fig. 1a and S5, ESI[†]). We attempted to maintain the thermodynamic process in order to get well-ordered **PDI** self-assemblies by slowing the evaporation of the solvent. As shown in Fig. 1b and S6 (ESI[†]), the cross section enlarged to $1 \times 1 \mu\text{m}$ while the length was more than $40 \mu\text{m}$. A further slow growing process over several days, in 1 : 1 chloroform/acetonitrile solvent resulted the formation of micron-scale rod structure. Most of the rods are more than $100 \mu\text{m}$ in length, with a $5 \times 5 \mu\text{m}$ cross section (Fig. 1c, d and S7, ESI[†]). These results indicated that the rod like structure undergo a bottom-up growing process from molecular to well-ordered rod structure. Powder X-ray powder diffraction (PXRD) measurements of **PDI** showed a typical π - π stacking peak (with d -spacing 3.5 \AA), which indicated that the building blocks form a well-ordered crystalline arrangement¹⁹ (Fig. S8, ESI[†]). According to the reported literature,²⁰ the strong π - π interaction between the **PDI** scaffolds played the key role in the formation of the rod like aggregates. The anisotropy character of the extended rod-like aggregate was examined by polarized light microscopy. The alternate appearance of birefringence when the rod was aligned at 45° to the polarizer implied that the optical axis is along the orientation of π - π stacking (Fig. 1e and f). No significant birefringence was observed when the agglomerates were examined under crossed polarizer

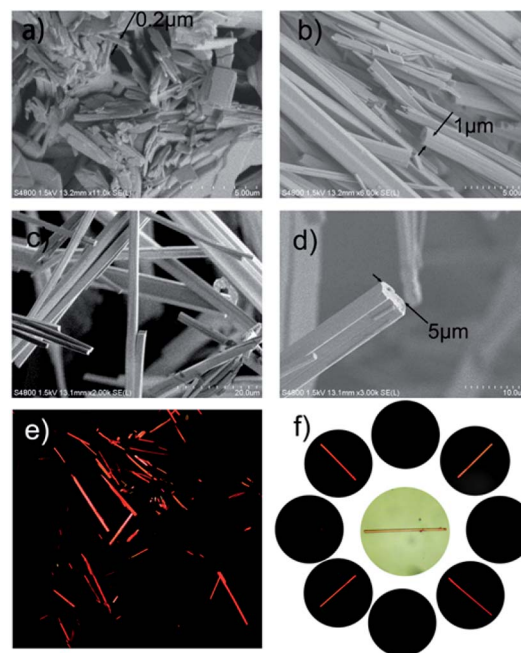


Fig. 1 SEM images of **PDI**. (a) The solvent was evaporated in vacuum, (b) the solvent was naturally evaporated in several hours, (c) and (d) the solvent was slowly evaporated in several days, (e) **PDI** rods under cross-polarized microscope, (f) consecutive rotation of a single **PDI** rod showed alternate appearance of birefringence as the rod was aligned at 45° to the polarizer.



conditions (Fig. S10, ESI[†]), indicating the randomly embedded submicrometer aggregate reduced the overall anisotropy of the agglomerates.¹⁶

2.2 Construction of anisotropic fluorescent nanofiber

However, our goal was to fabricate the prepared bricks (**PDI** assemblies) into second level structure in a high oriented way which would enable more expedient construction of integrated 1D nanofibers material. Since uniaxial optical property has a potential influence in the sensitivity of optical sensors, so the main problem to be faced was to keep the **PDI** assemblies in an ordered way while shorten the time of self-assembly process. An easy one-step method was developed to arrange the prepared aggregation in a neat second level row. The **PDI** aggregation was fabricated in a nanoscale limited space during the electrospinning process and then immediately encapsulated in the electrospun PVDF fiber. The whole process was limited in several milliseconds. As we had already mentioned that the millisecond time-scale could control the aggregate in a nanoscale size by shortening (disrupt) the assembly process meanwhile the micro/nano-scaled diameter provides axial confining enclosure (interspace) for arranging the assemblies in an order way.

The microscope images of the electrospun PVDF fiber (**PDI@PVDF**) are presented in Fig. 2. The characteristic SEM image (Fig. 2a and b) reveals the randomly-arranged fibers were gathered by a flat collector. From these images we can see that

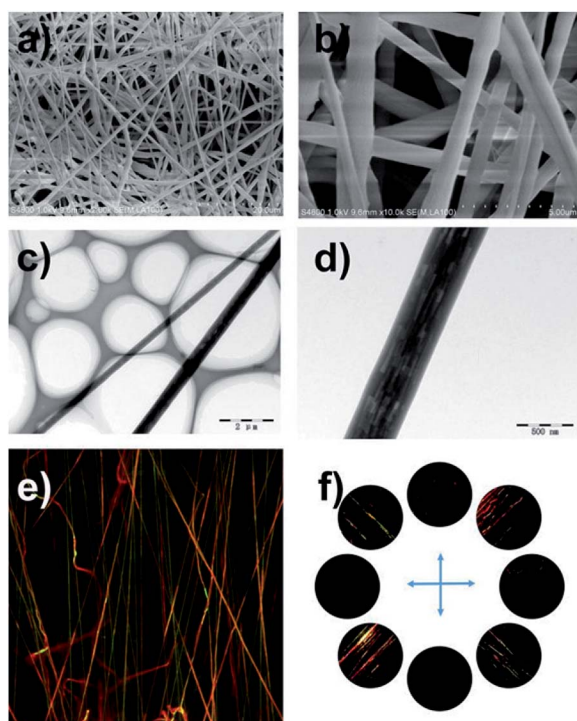


Fig. 2 (a) SEM (scale bar = 20 μm), (b) enlarged SEM (scale bar = 5 μm), (c) TEM (scale bar = 2 μm), (d) enlarged TEM (scale bar = 500 nm), (e) confocal microscopy and (f) cross-polarized microscope images of **PDI@PVDF** fiber formed from **PDI** and PVDF.

the diameters of these fibers (0.5–1 μm) are comparable to or less than the cross section of the rod-like **PDI** aggregate as shown in Fig. 1. The typical TEM images (Fig. 2c and d) show that the **PDI** aggregates are confined in the PVDF matrix, of which the longitudinal axis paralleled to the fiber axis. An aligned fibers was obtained by using a rotating cylindrical mandrel collector as shown in Fig. 2e and f. The different emission color of the fiber indicated that the confined formation of the **PDI** aggregate in the PVDF fiber shared the same hierarchical assembly process which had been discussed above. The hierarchical **PDI** aggregate with different emission color were frozen in the PVDF matrix during the fiber drawing process. The anisotropy character of the **PDI@PVDF** fibers depends on not only the orientation of the fibers, but also the orientation of **PDI** assemblies in it. The approximate paralleled fibers were used in the fluorescent microscopy examination. When rotating the fibers under crossed polarization, the same result was obtained in Fig. 2f implied that the optical axis is along the fiber axis.²¹ The anisotropy characters of both pure **PDI** assemblies (Fig. 1f) and **PDI@PVDF** fiber (Fig. 2f) further confirmed the longitudinal axis of the **PDI** assemblies paralleled to the **PDI@PVDF** fiber axis.²¹

2.3 Optical sensing applications

As previously reported, **PDI** nanofibers with enhanced long-range exciton migration property rising from intermolecular π -electronic coupling can act as molecular wires for the fluorescence sensing of organic vapor.^{4–6} The prepared 1D **PDI@PVDF** fibril film (Fig. S11e and f, ESI[†]) in which the **PDI** assemblies was arranged along the long axis of the fiber, may also worked in a “molecular wire” mode. For ease of comparison, herein, we also chose the frequently-used aniline as analyte molecules to investigate the amplified quenching behavior.

The **PDI@PVDF** film was fixed on a special designed scaffold then covered with a quartz cell (Fig. S12, ESI[†]). A 365 nm LED lamp was used to excite the film, and the fluorescence variation

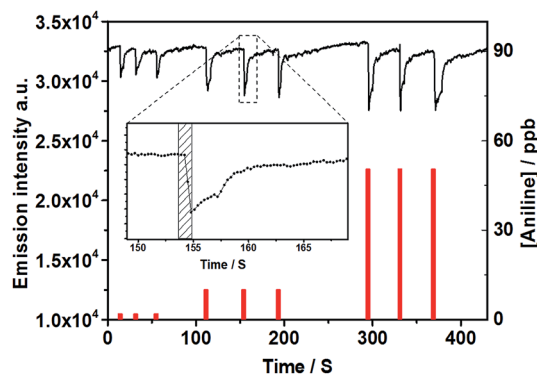


Fig. 3 Time-resolved fluorescence intensity (black line) of a **PDI@PVDF** fibril film upon consecutive exposures to various concentrations of aniline vapors. (1 s exposures, concentrations indicated by the red columns). Inset: fluorescence change upon exposure to 10 ppb aniline, and the filled rectangle indicated the exposure times (the emission intensity at 630 nm was measured for every 300 ms, λ_{ex} = 365 nm).



in the presence of organic amines (aniline) was continuously monitored by a spectrometer. As Fig. 3 showed, the fluorescence emission was significantly quenched by increasing aniline vapor concentration. Comparing with previous results, although the **PDI** was not attentional designed just like Liu⁵ and Fang⁶ reported, the **PDI@PVDF** fibril film still shown unexpected high sensitivity for aniline vapor. Concentration of aniline low to 0.4–50 ppb could induce observable fluorescence quenching (Fig. 3). Increased concentration up to 1250 ppb did result in further significant quenching (Fig. S13, ESI[†]). As an important factor for the practical use of a sensing film (Fig. S11e and f, ESI[†]), reversibility of the **PDI@PVDF** fibril film was investigated in detail. As shown in Fig. 3, the emission intensity of the film was monitored every 0.3 s. After being exposed to various concentrations of aniline vapors (Fig. 3 inset), the emission immediately quenched in less than 1 second. Additionally, the fluorescence quenching induced by amines was highly reversible. Upon exposed to 50 ppb aniline, the quenched fluorescence of the film could fully recover by re-exposing it to an ambient air-flow for only 20 seconds. However, the recovering time after exposure do has dependence on the aniline concentration. As shown in Fig. S13 (ESI[†]), the recovering time extended to 40 seconds when the concentration of aniline vapor reached 1250 ppb. This shorted response time for both quenching and recovering, in contrast to the other reversible **PDI** based sensor systems, in which the recovering time usually need 5 to 30 minutes due to the stable chemical binding between amines and the sensing material,^{15,22} indicated excellent reversibility of our hierarchical nanofiber.

Since the sensing behavior of **PDI@PVDF** fiber is mainly determined by **PDI** self-assembly, the fiber shows the same selectivity to organic amine. Common solvents like chloroform, toluene and methanol have little effect on the fluorescence emission of the film (Fig. S14, ESI[†]). Furthermore, the sensing behavior of **PDI@PVDF** fiber to different types of organic amines was investigated. As shown in Fig S14 (ESI[†]), the quenching efficiency induced by aromatic amines is higher than that induced by aliphatic amines, which might be attributed to the stronger electron-donating ability of the conjugated structure of aromatic amines. Additionally 4-methylaniline exhibited stronger quenching ability, while 4-nitroaniline did not cause significant quenching indicating that strong electron-withdrawing group on aniline might significantly reduce the electron cloud density of amino group and weaken the electron-donating ability of amino group. The significant quenching observed with phenol is likely due to its strong reducing power.²² Without specific modification of the molecular structure of **PDI**, it is difficult to distinguish phenolic compounds from organic amines only by fluorescence quenching. The static measurement and the fluorescence lifetime measurement has been performed in solution state to validate the quenching mechanism (Fig. S15 and S16, ESI[†]). According to the previous literature,¹⁵ present results (Fig. S17, ESI[†]) indicate that the fluorescence emission of **PDI@PVDF** fiber is quenched by both complex formation with aniline as well as by collision with aniline.

The rapid response behaviour of **PDI@PVDF** fibril film towards aniline can be attributed to the hierarchical assembly strategy. **PDI** firstly self-assembles into to well-ordered rod structure driven by the strong π - π interaction, resulting the sensing behaviour to aniline which depends on the photoinduced electron transfer from aniline to **PDI** probe.⁶ According to the literature,^{1b} the long-range molecular arrangement leads to 1D enhanced exciton migration (*via* intermolecular π -electronic coupling) along the long axis of the assemblies, enabling amplified fluorescence quenching by the surface adsorbed fewer aniline molecules (quencher molecules). Subsequent electrospinning the fluorescent **PDI@PVDF** into nanofibers not only maintain the anisotropy of **PDI** in the thin film, but also the porosity between nanofibers facilitate the diffusion of gaseous analyte molecules throughout the film matrix, thus leading to instant capture (and accumulation) or release of the vapor species. To verify our hypothesis, control experiments were carried out. We prepared dense thin film *via* physical doping **PDI** into PVDF matrix (denoted as **PDI**∩**PVDF**). The fluorescence emission of **PDI**∩**PVDF** and pure **PDI** molecular lamellae was also significantly quenched by aniline vapor. Compare to **PDI@PVDF** fibril film, the even higher quenching efficiency might be attributed to the much longer long-range molecular arrangement. Time-resolved fluorescence experiment showed that the recovery time of **PDI**∩**PVDF** and pure **PDI** molecular lamellae towards aniline vapor are 6 min and 19 min (Fig. S18, ESI[†]), respectively, both significant longer than **PDI@PVDF** fibril film. These results demonstrate that electrospinning assistant is a versatile and effective strategy to simultaneously maintain the anisotropy of 1D nanomaterials and accelerate the speed of their response to external stimulus. The ultra-sensitivity, the rapid sensing response, excellent reversibility along with the simple and economical preparation make the nanofibril **PDI@PVDF** film an ideal sensing material for designing trace amine onsite monitoring device.

3. Conclusions

We prepared a simple electrospun fiber for organic amines sensing, in which PVDF was chosen as mortar, meanwhile a simple perylene bisimide derivative **PDI** was used as the functional bricks. The well-defined **PDI** assemblies were embedded in the PVDF fiber in a highly oriented way along the axis of the fiber. Profiting from the structural integrity and geometric uniformity of the **PDI** assemblies in nanoscale and the 3D continuous, the fibroid electrospun **PDI@PVDF** film with micrometer scaled mesh-like porous secondary structure not only maintain high sensitivity for aniline vapour but also exhibit un-expected short response time for both quenching and recovering. Compared to fragile intrinsic **PDI** rod-like assemblies, the electrospun fiber **PDI@PVDF** film also shows greater flexibility originating from the soft PVDF scaffold. This report may provide an innovative approach to construction 1D fibers with desired size and morphology *via* appropriate choice of various lime mortar (alternative polymer) and brick (assembly blocks).



Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors acknowledge support for this work from the National Natural Science Foundation of China (21402237).

References

- (a) G. Yuan, H. Jiang, L. Zhang, Y. Liu and Y. Cui, *Coord. Chem. Rev.*, 2019, **378**, 483–499; (b) F. Wurthner, C. R. Moller, B. Fimmel, S. Ogi, P. Leowanawat and D. Schmidt, *Chem. Rev.*, 2016, **116**, 962–1052; (c) X. Hu, X. Wu, S. Wang, D. Chen, W. Xia, C. Lin, Y. Pan and L. Wang, *Polym. Chem.*, 2013, **4**, 4292–4297; (d) Z. Li, Z. Hou, H. Fan and H. Li, *Adv. Funct. Mater.*, 2017, **27**, 1604379.
- (a) Y. Chen and Y. Liu, *Adv. Mater.*, 2015, **27**, 5403–5409; (b) L. Zang, Y. Che and J. S. Moore, *Acc. Chem. Res.*, 2008, **41**, 1596–1608; (c) Y. J. Kim, D. Y. Kim, J. H. Lee, C. Nah, J. H. Lee, M. H. Lee, H. Y. Kim, S. W. Kuo, S. Shin and K. U. Jeong, *J. Mater. Chem.*, 2012, **22**, 13477–13480; (d) L. Liu, X. Chen, K. Liu, M. He, G. Wang, X. Chang and Y. Fang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 21364–21372; (e) X. Chang, Z. Zhou, C. Shang, G. Wang, Z. Wang, Y. Qi, Z. Li, H. Wang, L. Cao, X. Li, Y. Fang and P. J. Stang, *J. Am. Chem. Soc.*, 2019, **141**, 1757–1765.
- Y. Zhao and L. Jiang, *Adv. Mater.*, 2009, **21**, 3621–3638.
- (a) J. F. Fennell Jr, S. F. Liu, J. M. Azzarelli, J. G. Weis, S. Rochat, K. A. Mirica, J. B. Ravensbaek and T. M. Swager, *Angew. Chem., Int. Ed. Engl.*, 2016, **55**, 1266–1281; (b) R. Agarwal, *Small*, 2008, **4**, 1872–1893.
- (a) P. Wang, Y. Wang and L. Tong, *Light: Sci. Appl.*, 2013, **2**, e102; (b) A. K. Feldman, M. L. Steigerwald, X. Guo and C. Nuckolls, *Acc. Chem. Res.*, 2008, **41**, 1731–1741; (c) Q. H. Cui, Y. S. Zhao and J. Yao, *Chem. Sci.*, 2014, **5**, 52–57; (d) S. Guo, Y. Song, Y. He, X. Hu and L. Wang, *Angew. Chem.*, 2018, **130**, 3217–3221.
- (a) S. Wang, M. Qiao, Z. Ye, D. Dou, M. Chen, Y. Peng, Y. Shi, X. Yang, L. Cui, J. Li, C. Li, B. Wei and W.-Y. Wong, *iScience*, 2018, **9**, 532–541; (b) T. S. Kang, A. P. Smith, B. E. Taylor and M. F. Durstock, *Nano Lett.*, 2009, **9**, 601–606.
- Y. Wang, J. He, C. Liu, W. H. Chong and H. Chen, *Angew. Chem., Int. Ed.*, 2015, **54**, 2022–2051.
- X. Ma, F. Li, Z. Xie, M. Xue, Z. Zheng and X. Zhang, *Soft Matter*, 2017, **13**, 3685–3689.
- J. Yong, F. Chen, Q. Yang, J. Huo and X. Hou, *Chem. Soc. Rev.*, 2017, **46**, 4168–4217.
- H. W. Liang, S. Liu and S. H. Yu, *Adv. Mater.*, 2010, **22**, 3925–3937.
- (a) C. L. Zhang and S. H. Yu, *Chem. Soc. Rev.*, 2014, **43**, 4423–4448; (b) S. Maity, K. A. Kozek, W. C. Wu, J. B. Tracy, J. R. Bochinski and L. I. Clarke, *Part. Part. Syst. Character.*, 2013, **30**, 193–202; (c) H. J. Kim, W. B. Jung, H. S. Jeong and H. T. Jung, *J. Mater. Chem. C*, 2017, **5**, 12241–12248.
- E. Schoolaert, R. Hoogenboom and K. De Clerck, *Adv. Funct. Mater.*, 2017, **27**, 1702646.
- K. Sano, Y. Ishida and T. Aida, *Angew. Chem., Int. Ed. Engl.*, 2018, **57**, 2532–2543.
- A. M. Kendhale, A. P. H. J. Schenning and M. G. Debije, *J. Mater. Chem. A*, 2013, **1**, 229–232.
- H. Peng, L. Ding, T. Liu, X. Chen, L. Li, S. Yin and Y. Fang, *Chem. - Asian J.*, 2012, **7**, 1576–1582.
- K. Balakrishnan, A. Datar, R. Oitker, H. Chen, J. M. Zuo and L. Zang, *J. Am. Chem. Soc.*, 2005, **127**, 10496–10497.
- S. Prasanthkumar, S. Ghosh, V. C. Nair, A. Saeki, S. Seki and A. Ajayaghosh, *Angew. Chem., Int. Ed. Engl.*, 2015, **54**, 946–950.
- (a) Y. Liu, K. R. Wang, D.-S. Guo and B. P. Jiang, *Adv. Funct. Mater.*, 2009, **19**, 2230–2235; (b) D. J. Yang, I. Kamienchick, D. Y. Youn, A. Rothschild and I. D. Kim, *Adv. Funct. Mater.*, 2010, **20**, 4258–4264; (c) K. Liu, C. Shang, Z. Wang, Y. Qi, R. Miao, K. Liu, T. Liu and Y. Fang, *Nat. Commun.*, 2018, **9**, 1695.
- N. Wu, Y. Zhang, C. Wang, P. M. Slattum, X. Yang and L. Zang, *J. Phys. Chem. Lett.*, 2017, **8**, 292–298.
- K. Balakrishnan, A. Datar, T. Naddo, J. L. Huang, R. Oitker, M. Yen, J. C. Zhao and L. Zang, *J. Am. Chem. Soc.*, 2006, **128**, 7390–7398.
- Y. J. Kim, D. Y. Kim, J. H. Lee, C. Nah, J. H. Lee, M. H. Lee, H. Y. Kim, S. W. Kuo, S. Shin and K. U. Jeong, *J. Mater. Chem.*, 2012, **22**, 13477–13480.
- Y. Che, X. Yang, S. Loser and L. Zang, *Nano Lett.*, 2008, **8**(8), 2219–2223.

