

MINIREVIEW

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Synthesis and application of polypyrrole nanofibers: a review

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State-of-the-art polypyrrole nanofiber-based nanoarchitectonics can be generally fabricated by electrospinning, interfacial polymerization and reactive template methods. Even though analogous nanofiber morphologies and nanofibrous network architectures can be obtained by these methods, the structural details and structural complexities may alter significantly as different synthesis methods are applied. For the electrospinning technique, on one hand, nanofibers can be directly obtained by spinning polypyrrole-containing dope solutions; on the other, the electrospun nanofiber mats can be used as templates to direct the nanofiber formation; a two-step fabrication process, including the electrospinning of polymer nanofiber mats and deposition of polypyrrole on the polymer nanofibers' surface, is generally employed. By tuning the electrospinning parameters, the composition, diameter, morphology, and alignment of the as-obtained electrospun nanofiber mat can be effectively controlled, which may allow the fabrication of polypyrrole nanofibers with sophisticated nanostructures and nanoarchitectures. Interfacial polymerization is capable of generating polypyrrole nanofibers without templates. It is speculated that the protonation and re-orientation of polypyrrole at the oil–water interface may decoil the polymer chains and transform them into more extended conformations, while the charged polymer chains more easily diffuse into the water phase and form a stable dispersion. Different from electrospinning, the reactive templates may drive the formation of polypyrrole nanofibers through either redox or protonation mechanisms.

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1. Introduction

solution.¹⁶ During the electrospinning process, the dope solution is loaded into a reservoir equipped with a blunt-needle-type outlet. To start electrospinning, the dope solution is ejected through the outlet at a constant flow rate by applying pressure and a voltage of up to 50 kV is applied to the needle. As a result of the applied voltage, charges will accumulate on the surface of the liquid meniscus at the needle tip, ejecting a continuous liquid jet from the meniscus, forming a hydrodynamic structure called a Taylor cone.¹⁷ The liquid jet will move towards the collector, which is grounded or negatively charged, and placed several centimeters away from the needle tip. During the movement, the liquid jet will split, stretch, and solidify, forming nanofibers that randomly deposit on the surface of the collector; a schematic illustration of the electrospinning process of nanofibers is shown in Scheme 1. To obtain nanofiber mats with desired morphologies, structures, pore size distributions, sizes, and thicknesses, the electrospinning parameters, including the viscosity, surface tension, applied voltage, flow rate, needle size, needle/collector distance, collector dimensions, temperature and relative humidity, should be controlled carefully. Electrospinning might be the most versatile technique for fabricating nanofibers with different compositions, morphologies, structures and functions. It has been used extensively to fabricate different types of polypyrrole nanofibers, and the polypyrrole nanofiber membranes obtained from electrospinning have been used in a wide range of applications, *e.g.*, cell culture, tissue engineering, cancer therapy, biosensors, and supercapacitors.^{18–22}

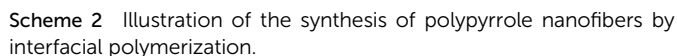
2. Electrospinning

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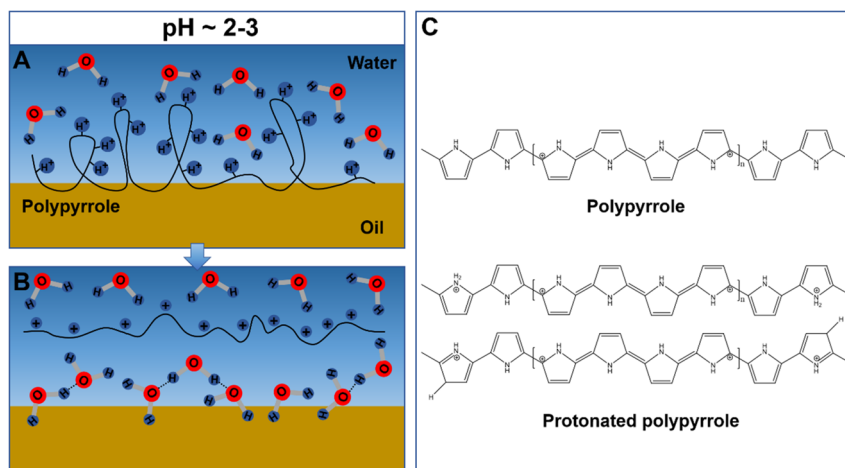
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Even though electrospinning has been proven to be one of the most versatile techniques to generate polymer nanofiber membranes and composites, it may require precise control over a wide range of processing parameters to achieve continuous and bead-free production of nanofibers, and the tuning of the parameters, *e.g.*, applied voltage, viscosity, temperature and relative humidity, is tedious and empirical, which may lower the efficiency of the fabrication process. High applied voltages and organic solvents are used during the electrospinning process, which may make it energy-consuming and environment-polluting. Nanofibers obtained from electrospinning may have a relatively wide range of size distribution and it is hard to obtain monodisperse nanofibers.

Polypyrrole nanofibers can be directly obtained from the free-radical polymerization of pyrrole through interfacial polycondensation. In a typical process, pyrrole is dissolved in an organic phase, *e.g.*, xylene, carbon disulfide, carbon tetrachloride, chloroform, methylene chloride, or toluene, while a strong acid (*e.g.*, 1 M hydrochloric acid) and a persulfate-type initiator [*e.g.*, ammonium persulfate (APS)] are mixed homogeneously to form the aqueous phase. To initiate the interfacial polymerization process, the water phase is carefully transferred to the top of the organic phase in a container, or *vice versa*, depending on the difference in density between these two phases. Due to the immiscibility of the two phases, an oil–water interface would form and the formation of the polypyrrole nanofibers may occur by the diffusion of the pyrrole monomer from the organic solvent to the aqueous phase *via* the oil–water interface. The interfacial polymerization can proceed at room temperature and the reaction occurs overnight to obtain polypyrrole nanofibers with sufficient degrees of polymerization and well-developed nanofibrous morphologies. Moreover, the as-obtained nanofibers can be assembled into a nanofibrous



Compared to the fabrication of polypyrrole nanofibers by using electrospinning, where the electrospun nanofibers can be used as the hard template, no specific templates are required for the fabrication of polypyrrole nanofibers *via* interfacial polymerization. It is speculated that the formation of the nanofibrous morphology is closely related to the adsorption, orientation and desorption of the polypyrrole molecules at the oil-water interface and is highly pH-dependent. The pH-dependent adsorption/desorption process of polyelectrolytes has been investigated by D. K. Beaman *et al.*, who showed the change in pH can significantly alter the conformation of the polyelectrolytes, as a result of the increased charge density in the polymer backbone; the charge accumulation in the backbones would induce the intra/inter-molecular repulsion of the polymer, driving the swell and de-coil of the polymer chains, forming more extended and linear conformations; the charges in the polymer backbone can promote the ionization of the neighboring groups and reduce the free energy of desorption, and the entire polymer chain may move to the bulk water when a substantial portion of it is solvated and the free energy of desorption is sufficiently low.²⁸ The formation of the polypyrrole nanofibers through interfacial polymerization can also be explained by the pH-dependent adsorption/desorption of the polypyrrole chains at the oil-water interface. Briefly speaking, when the water phase containing acids and initiators is spread on top of the oil phase containing pyrrole, the pyrrole monomers adsorbed at the oil-water interface would be oxidized by the persulfate ions, forming pyrrole radicals and polymerizing through free radical polymerization. Simultaneously, the as-obtained polypyrrole chains would be protonated by the protons adsorbed at the oil-water interface. Q. Pei *et al.* reported that polypyrrole can be partially protonated in an aqueous solution with pH around 2–3,²⁹ which is consistent with the pH value of the aqueous phase used for the interfacial polymerization. As a result of the electrostatic repulsion, the protonated polypyrrole chains would adopt a more linear and extended conformation instead of the random coil conformation, which may promote the formation of nanofibers; the charges distributed along the protonated polypyrrole chains can reduce the free energy of desorption and promote the solvation of the as-formed nanofibers in the bulk water phase. Short and



Scheme 3 pH-dependent adsorption and desorption of polypyrrole at the oil–water interface. Polypyrrole is firstly protonated by the protons (H^+) at pH ~ 2–3 (A), and subsequently adopts an extended conformation as a result of the electrostatic interaction, and desorbs from the oil–water interface due to a reduction in free energy (B). (C) The chemical structures of the pristine polypyrrole and protonated polypyrrole.

curved nanofibers with small diameters (10–80 nm) can be formed by using interfacial polymerization.^{30–32} However, by harmonizing the electrochemical and interfacial polymerization techniques, polypyrrole nanofiber arrays with controllable nanofiber lengths can be obtained.³³ A schematic illustration of the pH-dependent adsorption/desorption process of polypyrrole at the oil–water interface is shown in Scheme 3.

The addition of surface-active reagents, *e.g.*, surfactants, during the interfacial polymerization may significantly alter the morphology of the polypyrrole nanofibers. Recently, Z. Hou *et al.* investigated the effect of the hydrophilic–lipophilic balance (HLB) value on the morphology of polypyrrole synthesized by interfacial polymerization.³⁴ The HLB value was tuned by mixing different proportions of Span80 and OP10 to make a mixture surfactant. The mixture surfactant and $FeCl_3$ were dissolved in water to make the aqueous phase, while pyrrole was

dissolved in chloroform to make the oil phase. As the HLB value of the mixture surfactant increased from 4.3 to 14.5, the morphology of the as-obtained polypyrrole changed from ruptured vesicle-like (Fig. 1A and B) to network-like (Fig. 1C and D) and to coral-like (Fig. 1E and F), respectively. The formation of the network-like structure at HLB = 10.4 indicated that the mixture surfactant in this case was more effective in de-coiling and stabilizing the polypyrrole chains at the interface. However, the formation of a primitive network-like structure instead of a well-defined nanofibrous interpenetrating network in the absence of a strong acid indicates that the protonation and extension of the polypyrrole chains under acidic conditions is vital in promoting the nanofiber formation.

The morphology of polypyrrole synthesized by interfacial polymerization is very sensitive to the synthesis conditions. H. Albaris *et al.* reported the use of interfacial polymerization to

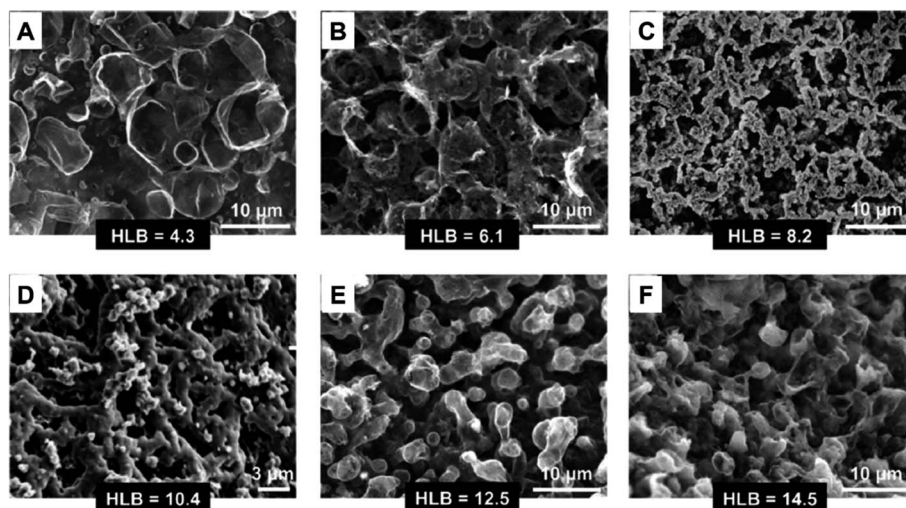
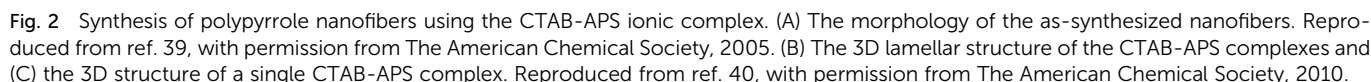


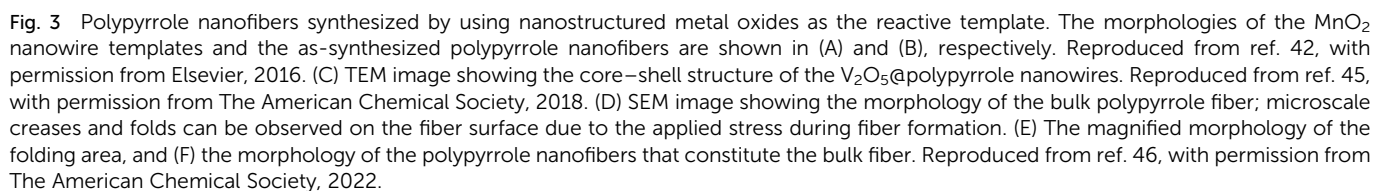
Fig. 1 Changes in the morphology of polypyrrole synthesized by interfacial polymerization with the addition of mixture surfactants with different HLB values. The morphologies of the as-synthesized polypyrrole alter from ruptured vesicle-like (A and B) to network-like (C and D), and eventually to coral-like (E and F) as the HLB value of the surfactant increased. Reproduced from ref. 34, with permission from Elsevier, 2018.



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morphologies, the nanofibers showed the best electrochemical properties for energy storage applications, possibly due to their ability to assemble into interconnected network architectures. Other than nanofibers solely composed of polypyrrole, core-shell nanofibers composed of a metal oxide core and a polypyrrole shell can also be obtained by using the reactive template. J. Wang *et al.* reported the fabrication of V_2O_5 @polypyrrole core-shell nanowires by using V_2O_5 nanowires as the reactive template.⁴⁵ The V_2O_5 nanowires were firstly synthesized by a hydrothermal method and then dispersed in an aqueous solution of sodium dodecylbenzenesulfonate (NaDBS). The pH of the dispersion was adjusted to 1 using diluted H_2SO_4 and the temperature was maintained at 0–5 °C by using an icy bath. Subsequently the polymerization process was started by adding pyrrole, which was oxidized by the V_2O_5



nanowire templates and formed a polypyrrole layer on the surface of the nanowires. The TEM image of the V_2O_5 @-polypyrrole core-shell nanowires is shown in Fig. 3C, where the crystalline V_2O_5 cores and the amorphous polypyrrole shells can be clearly distinguished. To ensure the formation of V_2O_5 @-polypyrrole core-shell nanowires and avoid the excess displacement of the V_2O_5 cores by pyrrole, the molar ratio between V_2O_5 and pyrrole was set at around 1 : 0.66. Compared to the samples consisting solely of V_2O_5 nanowires or polypyrrole, the V_2O_5 @polypyrrole core-shell nanowires showed remarkable improvements in specific capacitance, rate performance and cycling stability. Nanofiber dispersions and nanofibrous membranes composed of randomly oriented polypyrrole nanofibers can be obtained by the reactive template methods in general, while new nanoarchitectures are rarely reported. By adding low-molecular-weight polyethylene glycol (PEG) into the reactive template of V_2O_5 nanofiber sol, bulk polypyrrole fibers composed of polypyrrole nanofibers assembled along the fiber long axis can be synthesized in a one-step manner.⁴⁶ This new nanoarchitecture of polypyrrole typically showed diameters in the range of 10–50 μm , while their lengths can reach several centimeters. The surface of the bulk polypyrrole fibers is not smooth, and is roughened by intensive creases and folds that can span over several tens of microns (Fig. 3D). The SEM image with higher magnification shows the detailed morphology of the creases and folds, revealing a hierarchical micro/nano-structure composed of stacking layers of polypyrrole nanofibers (Fig. 3E). Moreover, the micro-creases and folds formed by the mismatch between the layers are observed to be partially aligned with the long axis of the fiber, which can be attributed to the shear stress applied on the V_2O_5 nanofiber sol during the synthesis process. The surface nanostructure of the bulk polypyrrole fiber is shown in Fig. 3F, as a dense layer of interwoven polypyrrole nanofibers with an average diameter of 63 nm and a standard deviation of ± 7 nm can be observed. Compared to the polypyrrole nanofibers synthesized solely by V_2O_5 nanofiber sol and polypyrrole granules synthesized without adding the V_2O_5 sol, the bulk polypyrrole fibers showed enhanced electronic properties, *i.e.*, a larger band gap, a higher bipolaron density, and a higher degree of charge carrier delocalization, which can be ascribed to the strong hydrogen bonding and charge transfer interaction between the PEG molecules and polypyrrole.

Different from the reactive templates that drive the nanofiber formation through a redox mechanism, the nanofibrous templates that are able to protonate pyrrole monomers can also induce the formation of polypyrrole nanofibers. Pyrrole radicals can form by the protonation process, which can initiate the polymerization of pyrrole through a trimerization mechanism (Fig. 4). Polypyrrole nanofibers with unique helicity can be synthesized by using chiral amino acids as the template. For example, A. Xie *et al.* reported the use of left-handed and right-handed *N*-myristoyl-glutamic acids (L/R-MGA) to synthesize single-handed helical polypyrrole nanotubes with diameters of around 80 nm in an ethanol/water solution.⁴⁷ Polypyrrole nanotubes with either a left-handed helical structure or a right-handed helical structure can be obtained by using L-MGA or R-

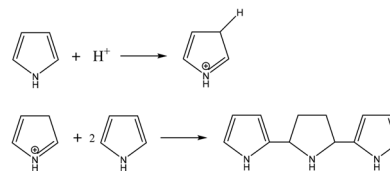


Fig. 4 Protonation and trimerization processes of pyrrole in acids.

MGA, respectively. Helical polypyrrole nanofibers with diameters of around 40 nm can also be obtained by replacing the MGA templates with a gel dispersion of *N*-myristoyl-L-diglutamic acid (L-MDGA). When being used as the electromagnetic interference (EMI) shielding materials, the helical nanostructured polypyrrole showed both high shielding efficiency and wide absorbing frequency range, while the highest absorption width of 7.04 GHz was obtained for the helical polypyrrole nanofibers in the frequency range between 10.6 and 17.64 GHz. By adding different acids into the polymerization solution of the helical polypyrrole nanotubes, M. Sun *et al.* were able to synthesize hollow polypyrrole nanofibers with tunable microstructures, which can be used for EMI shielding.⁴⁸

On the other hand, electrospun nanofiber mats can also be used as either non-reactive or reactive templates to synthesize polypyrrole nanofibers. For example, R. Elashnikov *et al.* prepared polypyrrole nanofiber membranes for cell culture and stimulation by using an electrospun cellulose acetate butyrate (CAB) nanofiber mat as the substrate;⁴⁹ SEM images of the as-obtained CAB nanofiber mat are shown in Fig. 5A and B. Polypyrrole was then deposited onto the CAB nanofiber mat by chronopotentiometry. Both randomly oriented and uniaxially aligned polypyrrole nanofiber membranes can be obtained from the electrospinning process, by using either a single plate or parallel plates as the collector (Fig. 5C and D). The as-obtained polypyrrole nanofiber membranes can significantly promote the growth of the human neuroblastoma cells (SH-SY5Y) when electrical stimulation is applied, resulting in longer neurites with more branches. Compared to the randomly oriented nanofiber mat, the uniaxially aligned nanofiber mat can induce directional neurite growth, resulting in a higher average length for the as-grown neurites. Y. Zhou *et al.* reported the fabrication of a high-performance yarn-based strain sensor by using the polypyrrole nanofiber composites.⁵⁰ According to their method, a spandex yarn was pre-stretched and fixed in between two twisting rollers connected to a winding device. The graphene oxide (GO) doped polyacrylonitrile (PAN) nanofibers were then electrospun onto the spandex yarn and covered its surface. Subsequently the nanofiber coated yarn was twisted by the twisting rollers to entangle the nanofibers and the spandex yarn. After removing the stretching load, the nanofibers would contract to form a uniform wavy structure on the surface of the yarn, as shown in Fig. 5E and F. Polypyrrole was then deposited onto the surface of the PAN nanofiber-coated spandex yarn by a free radical polymerization process in the aqueous phase, forming a continuous conductive layer on the PAN nanofibers' surface, as shown in Fig. 5G and H. The as-obtained composite yarn was highly adaptable to the weaving process and showed



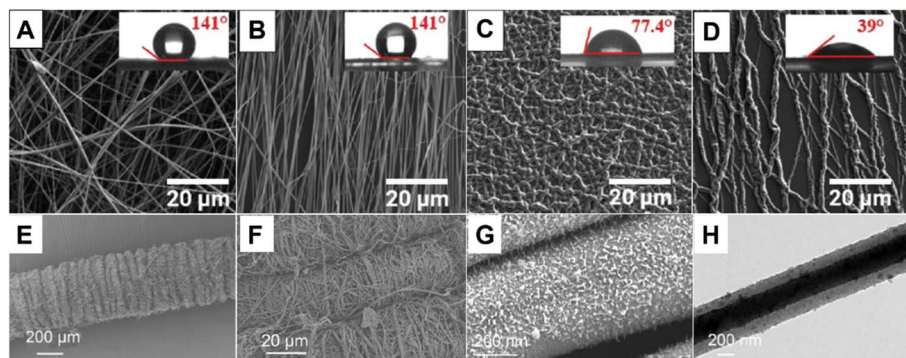


Fig. 5 SEM images of the electrospun CAB (A and B) and polypyrrole coated CAB nanofiber membranes (C and D) with either random (A and C) or uniaxially aligned (B and D) orientations. Insets show the water contact angles of the membranes, respectively. Reproduced from ref. 49, with permission from Royal Society of Chemistry, 2019. (E) SEM image of the polypyrrole/GO doped-PAN-nanofiber coated spandex yarn with a wavy surface texture. (F) The wavy surface texture is formed by the contraction of the nanofiber coating. (G and H) SEM and TEM images of the polypyrrole layer deposited on the GO doped PAN nanofiber surface. Reproduced from ref. 50, with permission from The American Chemical Society, 2022.

remarkable sensory properties when used as a strain sensor. Moreover, D. P. Bhattarai *et al.* reported the use of an electrospun polycaprolactone (PCL) nanofiber mat as a sacrificial template to fabricate hollow polypyrrole nanofibers.⁵¹ Polypyrrole was deposited onto the surface of the PCL nanofiber mat by free radical polymerization to form a PCL/polypyrrole core-shell membrane. Subsequently the PCL core was dissolved in dichloromethane, leaving hollow polypyrrole nanofibers. The hollow polypyrrole nanofibers showed no toxicity against breast cancer cells (MCF7) under ambient conditions, while a strong photothermal toxicity against the MCF7 cells can be obtained by irradiating the nanofibers with an 808 nm near-infrared (NIR) light source. Compared to the polypyrrole nanoparticles, the hollow nanofibers showed much higher drug-loading capacity, which can be attributed to the hollow-fiber microstructure. Upon the incorporation of oxidants, electrospun nanofibers can also be used as reactive templates to prepare polypyrrole nanofibers. X. Lin *et al.* reported the fabrication of an all-nanofiber-based piezoresistive sensor by using polypyrrole coated electrospun nanofibers.⁵² The sensor was composed of a polypyrrole/PVP/PAN nanofiber mat as the central layer, which was prepared by polymerizing pyrrole on the surface of the FeCl₃ containing electrospun PVP nanofibers. An electrospun thermoplastic polyurethane (TPU) nanofiber mat printed with interdigitated silver electrodes was used as the bottom layer, and an electrospun TPU nanofiber mat was used as the top layer. These layers were sandwiched to assemble the piezoresistive sensor, which showed excellent pressure sensing abilities against different body motions accompanied by good breathability and heat dissipation, making it a good candidate for electronic skins.

5. Structure–property correlations of the polypyrrole nanofibers

Polypyrrole nanofibers synthesized by using different methods have various structures and hence different properties.

Electrical conductivity is considered as one of the most important properties for conducting polymers, *e.g.*, polypyrrole, since the main applications of conducting polymers, such as in supercapacitors, organic solar cells, bioactive scaffolds, conductive fillers, and sensors, require high electrical conductivity to realize advanced functions and performance. The correlations between the electrical conductivities of the polypyrrole nanofibers and the corresponding synthesis methods in the literature are summarized in Table 1. From Table 1, it can be seen that the electrical conductivities of the nanofibers differ significantly based on the synthesis conditions, including the methods, morphologies, molar ratios, and dopants. By carefully investigating the detailed information and parameters in the literature, a few general rules can be deduced: (i) the electrical conductivity of the as-prepared nanofibers increases with the weight fraction of polypyrrole, especially for the electrospun nanofibers; (ii) the electrical conductivity of the polypyrrole nanofibers increases as the density of polypyrrole increases, *e.g.*, the conductivity of the solid polypyrrole nanofibers is generally higher than that of the hollow nanofibers or coatings; (iii) the conductivity of the as-prepared nanofibers depends on their composition, and the presence of other conductive components can enhance the electrical conductivity; (iv) organic dopants, such as benzenesulfonate anions, can induce higher electrical conductivity, which can be attributed to the electrostatic shielding effect originating from their large sizes; (v) protonation in the acidic solution may enhance the reactivity of pyrrole, resulting in the formation of polypyrrole with higher electrical conductivity. The unique nanofibrous morphology may provide anisotropic physical properties, larger surface area, and assemblability, which may significantly improve the efficiency of functionalization and nanoscale self-assembly. By taking advantage of the unique doping-induced properties of polypyrrole, *e.g.*, conductivity, structural change, and stimuli-responsiveness, important applications of polypyrrole nanofibers may be found across a wide range of research disciplines.





Table 1 Summary of the compositions, morphologies, and conductivities of the polypyrrole (PPy) nanofibers prepared by using different synthesis methods

Name	Synthesis method	PPy morphology	Molar ratio of Py to oxidant	Dopant	Conductivity	Testing method
PPy/chitosan/collagen nanofiber scaffold (15 wt% PPy) ⁴	Electrospinning	Blended in the electro-spun nanofibers; $d_{\text{avg}} = 83.7$ nm	0.95	Diethyl sulfosuccinate, SO_4^{2-}	$\sim 164 \times 10^{-3} \text{ S m}^{-1}$	4-Probe, sheet form
PPy nanofibers ⁸	Interfacial polymerization	Nanofiber bundles and networks; diameter range: 75–180 nm	7.27 (0.3 for FeCl_3)	HCl , FeCl_3 , p -TSA, CSA, PSSA	6.25×10^{-4} , 21×10^{-4} , 6×10^{-2} , 4×10^{-2} , $20 \times 10^{-4} \text{ S cm}^{-1}$	4-Probe, pressed pellet
PPy nanoribbons and nanowires ⁹	Reactive templating	Curved nanowire or nanoribbon networks; diameter: 20–65 nm	1	Br^- , SO_4^{2-}	$7.3 \times 10^{-3} \text{ S cm}^{-1}$	4-Probe, compressed pellet (10 MPa)
PPy/PVDF electrospun nanofibers ²⁰	Electrospinning	Blended in the electro-spun nanofibers; $d_{\text{avg}} = 325$ nm	0.5	FeCl_3	$\sim 1.6 \times 10^{-4} \text{ S cm}^{-1}$	LCR meter, nanofiber membrane
MnO_2 @PAA/PPy core-shell composite fibers ²²	Reactive templating	Granular coating, 50 nm thick	—	FeCl_3	$1.19 \times 10^{-3} \text{ S m}^{-1}$	2-Probe, fiber mat
PPy/PEO nanofibers ²⁶	Electrospinning	Blended in the electro-spun nanofibers; diameter range: 200–300 nm	—	Organic acids	4.9×10^{-8} to $1.2 \times 10^{-5} \text{ S cm}^{-1}$, depending on the PPy content	2-Probe, fiber mat
PPy/PVP nanofibers ²⁷	Electrospinning	Blended in the electro-spun nanofibers; $d_{\text{avg}} = 833$ nm	—	AQSA, FeCl_3	$5.22 \times 10^{-1} \text{ S cm}^{-1}$	4-Probe, nano-fiber mat
PPy nanofibers ³⁷	Interfacial polymerization	Nanofiber bundles	1	p -TSA	$3.3 \times 10^{-1} \text{ S cm}^{-1}$	4-Probe, pressed pellet
PPy nanofibers ³⁷	Interfacial polymerization	Nanofiber bundles	1	HCl	$9.7 \times 10^{-2} \text{ S cm}^{-1}$	4-Probe, pressed pellet
Polypyrrole nanofibers ³⁹	Reactive templating	Nonwoven mat of nanofibers	4.3	SO_4^{2-} , Cl^- , Br^-	$\sim 100 \text{ S cm}^{-1}$	Compressed pellet
PPy nanoclips ⁴⁰	Reactive templating	2D nanoclips	4	SO_4^{2-} , Cl^- , Br^-	$2-5 \text{ S cm}^{-1}$	4-Probe, compressed pellet
V_2O_5 @PPy core-shell nanowires ⁴⁵	Reactive templating	Nanocoating, 15 nm thick	0.65	DBS^- , SO_4^{2-}	—	—
PPy nanofiber assemblies ⁴⁶	Reactive templating	Uniaxially aligned bulk nanofiber assembly; $d_{\text{avg}} = 63$ nm	3.27	SO_4^{2-} , Cl^-	1.49 S m^{-1}	4-Probe, pressed pellet
PPy coated cellulose acetate butyrate nanofibers ⁴⁹	Non-reactive templating	Granular coating, 600 nm thick	—	H_2PO_4^-	$2.4 \times 10^{-5} \text{ S cm}^{-1}$	2-Probe, nano-fiber mat
Helical PPy nanofibers ⁵³	Non-reactive templating	Nanofibers with helical architectures; $d_{\text{avg}} = 113$ nm	1	SO_4^{2-}	1.83 S cm^{-1}	—
ANF-PVA-PPy hydrogel ⁵⁴	Non-reactive templating	Granular coating	—	FeCl_3	$\sim 80 \text{ S cm}^{-1}$	2-Probe, cuboid hydrogel

^a Abbreviations: d_{avg} (average diameter), p -TSA (p -toluene sulfonic acid), CSA (camphor sulfonic acid), PSSA (polystyrene sulfonic acid), PVDF (polyvinylidene fluoride), LCR (inductance, capacitance, and resistance), PAA (polyacrylic acid), PEO (polyethylene oxide), PVP (polyvinyl pyrrolidone), AQSA (anthraquinone-2-sulfonic acid sodium salt), DBS (dodecylbenzenesulfonate), ANF (aramid nanofibers), PVA (polyvinyl alcohol). ^b The conductivities of PPy nanofibers doped with HCl , FeCl_3 , p -TSA, CSA, and PSSA, respectively.

6. Emerging applications of polypyrrole nanofibers

6.1 Biomedical applications

Polypyrrole nanofibers have been researched for use in biomedical applications from medical diagnostics and drug delivery to tissue engineering and regeneration. Among them, drug delivery has long been researched, with the focus now shifting to PPy-based drug delivery systems with environmental stimuli-responsiveness including to pH and near-infrared (NIR).⁵⁵ Recently, synergistic therapy has also been developed for PPy-based drug delivery. For example, by encapsulating gold nanorods (GNRs) into polypyrrole (PPy) shells, photothermal and drug delivery performance for cancer therapy has been regulated.⁵⁶ Besides drug delivery, in recent years, a lot of research has been carried out on using PPy in medical diagnostics and tissue engineering. With the development of technologies such as toughening hydrogel, biodegradable polymers, biomimetics, and 3D printing, PPy nanofibers could find a wide range of applications in tissue engineering, as numerous studies have been carried out recently.^{57–60}

6.2 Energy storage and conversion

The high surface area, good electrical conductivity, and excellent electrochemical performance make PPy suitable for enhancing energy storage and conversion efficiency, as it is widely used in energy storage devices such as supercapacitors and batteries. PPy holds great promise as a cathode material for fuel cells to replace noble metal catalysts, as it is capable of combining advanced functions including high oxygen-reduction activity, charge density and good performance durability, as reported by R. Bashyam *et al.*⁶¹ The development of PPy-nanofiber synthesis technology will advance its application in fuel cells. Supercapacitors are another device in which PPy could play a great role. Flexible, stretchable, all-solid-state, fiber-shaped supercapacitors based on CNT-PPy fibers were fabricated, exhibiting high specific capacitance and excellent stretchable performance which could be used in wearable electronic devices.⁶² A triboelectric nanogenerator (TENG) made from PPy has also been reported, and it showed a competitive performance comparable to that of commercial pulse sources.⁶³ It is expected that the PPy nanofibers will find great applications in energy storage and conversion systems, realizing enormous energy storage capabilities and promoting the commercialization of wearable devices, sustainable fuel cells and so on.⁶⁴

6.3 Multifunctional actuators

With the increasing demand for integrated automatic machineries for repetitive industrial processes, it is of particular interest to develop advanced actuators for diverse applications ranging from robots and memory chips to micro-electromechanical systems (MES).⁶⁵ Compared to other materials, PPy actuators showed a high response strain rate, power density ($\sim 39 \text{ W kg}^{-1}$), and stability when applying a potential up to 7 V within 1 s.⁶⁶ Electrical stimuli with low applied voltages

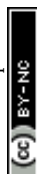
(electrochemical actuation) are the pros of the PPy based actuators, but challenges still exist, *e.g.*, how to simultaneously realize flexibility, durability, fast response and large stroke length. Hybridization of PPy with other carbon-based materials such as graphene and carbon nanotubes shows noticeable promise for constructing high-performance actuators.⁶⁷ Moreover, the incorporation of PPy into the wafer-scale nanoporous single-crystalline silicon membrane provided new perspectives for on-chip actuators with outstanding voltage-strain coupling and small operation voltages.⁶⁸

6.4 Sensors

Numerous studies have been conducted to develop PPy-based sensors, such as electrochemical, gas, and bio-sensors, since PPy is highly promising for commercial sensory applications as a result of its solution-processability, stimuli-responsiveness, and good environmental stability.⁶⁹ For biosensors, the analysis of selected DNA sequences and mutated genes associated with human disease is of significant importance. PPy may serve as a facile, high-performance and inexpensive functional material for fulfilling this goal. For example, a polypyrrole-based DNA chip has been designed and applied to the analysis of genomic DNA originating from cell lines and human colorectal samples, while practical features, such as low cost, high accuracy, and adaptability, can be realized.⁷⁰ Because of its high conductivity and electrochemical activity, PPy has been integrated with different substrates for collecting acute signals such as from genes,⁷¹ ammonia gas, thermal heating, NIR light,⁷² and volatile organic compounds.⁷³

7. Conclusions

Polypyrrole nanofibers with different compositions, nano-architectures and combinations can be synthesized by using electrospinning, interfacial polymerization, and solution-based template methods. For the electrospinning method, the electrospun nanofiber mats can be utilized as the substrate for polypyrrole deposition and as the templates to direct nanofiber formation. A variety of polypyrrole nanofiber-based nano-architectures can be facilely fabricated by the electrospinning technique, including core-shell nanofibers, hollow nanofibers, and uniaxially aligned nanofibers. Moreover, advanced fiber/nanofiber-based composites and devices can be fabricated by coating polypyrrole on the surface of the composite nanofibers prepared by the electrospinning of a multi-component dope solution or co-electrospinning of different dope solutions. Enhanced electrical conductivity, charge storage capacity, photothermal properties and biocompatibility can be provided by the polypyrrole coating layer, while enhanced mechanical properties, processability, and wearability can be provided by the electrospun nanofiber substrates. Given the diversity in composition, nanoarchitecture and combination, the polypyrrole nanofiber-based nanoarchitectonics show promising performance in a wide range of applications, *e.g.*, cell culture, neural stimulation, photothermal therapy, electronic skins, and strain sensors.



Compared to the electrospinning technique, interfacial polymerization can generate polypyrrole nanofibers without the presence of a template. It is speculated that a pH-dependent adsorption/desorption process at the oil–water interface may drive the formation of the nanofibers. As the polymer chains of polypyrrole are protonated, de-coiled, and desorbed into bulk water at the oil–water interface, the as-obtained polypyrrole may take up more extended and linear conformations, forming a stable dispersion in water due to the electrostatic repulsion. Consequently, curved polypyrrole nanofibers with small diameters are generally obtained by interfacial polymerization. On the other hand, the conformation of the polymer chains can be effectively tuned in the presence of the oil–water interface, which may enhance the stacking order of the polymer chains, resulting in a higher crystallinity and conductivity.

Compared to the interfacial polymerization and electrospinning technique, the synthesis of polypyrrole nanofibers by using reactive templates does not require complicated spinning devices, post-spinning deposition, or tedious interfacial adsorption/desorption processes, and the synthesis process can be completed in a simple, fast, and environment-friendly manner. Different from the non-reactive templates, the reactive templates can initiate the polymerization of pyrrole through redox mechanisms, driving the formation of polypyrrole on their surface and forming polypyrrole that duplicates their nanofibrous morphologies. Polypyrrole nanofibers with versatile morphologies and nanoarchitectures can be obtained by using the reactive templates, including the highly curved nanofibers, long and straight nanofibers, nanotubes, core-shell nanofibers, nanofiber urchins, and hierarchical polypyrrole bulk fibers, which show a wide range of applications in supercapacitors, lithium-ion batteries, organic electronics, and EMI shielding.

Conflicts of interest

The authors declare no conflicts of interest.

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