



Cite this: *CrystEngComm*, 2023, 25, 4503

Computation meets experiment: identification of highly efficient fibrillating peptides†

Lorenzo Sori, ^{‡a} Andrea Pizzi, ^{‡*a} Greta Bergamaschi, ^b Alessandro Gori, ^b Alfonso Gautieri, ^c Nicola Demitri, ^d Monica Soncini ^c and Pierangelo Metrangolo ^{*a}

Self-assembling peptides are of huge interest for biological, medical and nanotechnological applications. The enormous chemical variety that is available from the 20 amino acids offers potentially unlimited peptide sequences, but it is currently an issue to predict their supramolecular behavior in a reliable and cheap way. Herein we report a computational method to screen and forecast the aqueous self-assembly propensity of amyloidogenic pentapeptides. This method was found also as an interesting tool to predict peptide crystallinity, which may be of interest for the development of peptide based drugs.

Received 13th May 2023,
Accepted 3rd July 2023

DOI: 10.1039/d3ce00495c

rsc.li/crystengcomm

Introduction

Self-assembly processes lead molecular entities to organize themselves in particularly stable and functional supramolecular structures, moving from an original state of disorder to a subsequent one of order.¹ These are essential events that regulate various biological processes, which are mainly driven by noncovalent interactions.² The folding of nucleic acids and proteins is undoubtedly the best-known self-assembly phenomenon.³ In some specific conditions the resulting supramolecular structure does not exert a specific function; this is the case of amyloid fibers, which occur as a consequence of aberrant protein misfolding and aggregation and are considered the hallmark of several debilitating diseases like Alzheimer's (AD), Parkinson's (PD), Huntington's (HD) and type-II diabetes (T2D).^{4,5} Amyloids have a particularly defined hierarchical structure (Fig. 1) where the lamination of β -sheet layers forms the cross- β architecture, *i.e.* the core motif of supramolecular subunits (protofilaments) that group laterally and compose the finished fiber.

In addition to the interest closely related to these pathologies, the processes driving the formation of amyloids (fibrillation) are the subject of intense research in the fields of biomaterials and bioinspired materials, due to their intriguing mechanical properties.

In this context the study of self-assembling peptides to prepare functional nanostructures and materials⁶ is bringing new potential and opportunities to the fields of medicine and nanotechnology.⁷

The huge combinatorial possibilities offered by the chemical variety of natural and customized amino acids lead to almost unlimited peptide sequences, hence a wide range of achievable supramolecular morphologies. Unfortunately, the high potential of a system is strictly connected to its complexity. Indeed, the prediction and the control of the self-assembly of peptide-based systems are the factors that hamper their full exploitation.⁸

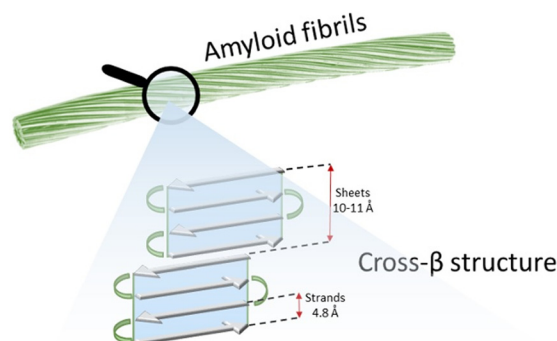


Fig. 1 Schematic representation of the structure of an amyloid fibril, formed by several protofilaments showing a cross- β structure.

^a Laboratory of Supramolecular and BioNano Materials (SupraBioNanoLab), Department of Chemistry, Materials, and Chemical Engineering "Giulio Natta", Politecnico di Milano, Via Luigi Mancinelli 7, 20131 Milan, Italy. E-mail: andrea.pizzi@polimi.it, pierangelo.metrangolo@polimi.it

^b Istituto di Scienze e Tecnologie Chimiche – National Research Council of Italy (SCITEC-CNR), 20131 Milan, Italy

^c Department of Electronics, Information and Bioengineering, Politecnico di Milano, 20131 Milan, Italy

^d Elettra – Sincrotrone Trieste, S.S. 14 Km 163.5 in Area Science Park, 34149 Basovizza – Trieste, Italy

† Electronic supplementary information (ESI) available: CCDC 2174444–2174447. For ESI and crystallographic data in CIF or other electronic format see DOI: <https://doi.org/10.1039/d3ce00495c>

‡ These authors equally contributed to this work.



Experimental validation of the used computational method revealed the importance of the nature of the N-term amino acid in the resulting supramolecular behavior, and led to the identification of theoretical conditions which may be related to the propensity of peptides in forming fibrils or crystalline self-assemblies.

Materials

Peptides were assembled using CTC resin (1.6 mmol g⁻¹ loading) by stepwise microwave-assisted Fmoc-SPPS on a Biotage ALSTRA Initiator+ peptide synthesizer, operating in a 0.1 mmol scale. Activation of entering Fmoc-protected amino acids (0.3 M solution in DMF) was performed using 0.5 M oxyma in DMF/0.5 M DIC in DMF (1 : 1 : 1 molar ratio), with a 5-equivalent excess over the initial resin loading. Coupling steps were performed for 20 minutes at 50 °C. Fmoc-deprotection steps were performed by treatment with a 20% piperidine solution in DMF at room temperature (1 × 10 min). Following each coupling or deprotection step, peptidyl-resin was washed with DMF (4 × 3.5 mL). Upon complete

chain assembly, resin was washed with DCM (5×3.5 mL) and gently dried under a nitrogen flow.

After cleavage from the resin, peptides were purified through RP-HPLC using a binary gradient of mobile phase A (A: 97.5% H₂O, 2.5% ACN, 0.7% TFA) and mobile phase B (30% water, 70% acetonitrile, 0.1% trifluoroacetic acid). The following chromatographic method was used: 0% B to 90% B in 45 min; flow rate, 14 mL min⁻¹. Pure RP-HPLC fractions (>95%) were combined and lyophilized.

More details on resin loading, cleavage from resin, work-up and analytical HPLC are available in the ESI.†

Rheometry

A KINEXUS Pro+ rheometer (MalvernPanalytical, UK) was used to measure the viscoelastic properties of the tested samples. Samples were pre-formed and directly transferred on the bottom rheometer plate. The upper geometry Cone 60 mm was lowered until it was in conformal contact with the top surface of the hydrogel, corresponding to gap distances of 1.0–1.5 mm. Temperature was controlled with a Peltier device and maintained at 25 °C. Each analysis was repeated at least 3 times, and representative measures are reported.

Coarse grain molecular dynamics

Pentapeptide coordinate files were created using VMD scripting tools and converted to coarse grain (CG) representation in the MARTINI force field (version 2.2) using martinize.py.³¹ As the secondary structure needs to be defined in the force field, the flag *-ss = EEE* was used, (E = extended β -sheet) leading to Qa, Nda and Qd beads for the backbone particles. This choice was made as β -sheet-like conformations are often observed in peptide nanostructures.³² Using the GROMACS code version 4.5.3,³³ a cubic box of $10 \times 10 \times 10$ nm³ containing 90 zwitterionic pentapeptides was created giving a peptide concentration of 0.1 mol L⁻¹ in explicit standard CG water, with side chains in their most prevalent charge state at pH 7. The box was energy minimized and equilibrated for 500 000 steps of 25 fs (equivalent to 50 ns of simulated time, considering the $4\times$ acceleration in CG dynamics), at a temperature of 303 K and a pressure of 1 bar. For the pentapeptides selected for further study, we also tested polarizable water (PW)³⁴ and longer simulations of 400 ns effective time. The MD simulations were performed on local workstations equipped with 12 CPUs and 1 NVIDIA Quadro RTX5000 GPU. For our models (≈ 9000 particles) the system provided a performance of ≈ 3 μ s per day. The overall simulated time is 8000×50 ns (400 μ s), thus requiring ≈ 130 days of computation.

Results and discussion

Numerous studies have focused on molecular self-assembly in the context of supramolecular materials using computational approaches.^{35,36} Inspired by these theoretical models, the self-assembly propensity of different well-known

peptide sequences such as NFGAIL,^{37,38} FF,^{39,40} FFF⁴¹ and KLVFFAE⁴² was studied.

Human calcitonin (hCT) is a 32 amino acid polypeptide hormone produced by thyroid C lymphocytes that is involved in calcium homeostasis.⁴³ This sequence, although much less studied than A β , displays a notable tendency to aggregate into fibrils; indeed, the formation of amyloids composed of hCT has long been associated with medullary thyroid carcinoma.⁴⁴

Some of us recently reported the high-resolution single crystal X-ray structure of an iodinated variant of DFNKF (a small fragment derived from hCT), which revealed the crucial role of aromatic interactions involving phenylalanines in the resulting β -sheet architecture.⁴⁵ Considering the intrinsic amyloid propensity of this sequence and the established importance of aromatic interactions, we selected DFNKF as a reference for our coarse-grained model.

Specifically, a wild-type peptide library was computed starting from the model sequence xFxxF, where x represents any of the 20 natural amino acids. Notably, the phenylalanine residues have been kept in the same position as in DFNKF, while all the possible peptide combinations (set of 20^3) were obtained through CG-MD upon varying the three x amino acids. For each computed pentapeptide, we estimated the aggregation propensity (AP). The AP value is defined as the ratio of solvent accessible surface areas (SASAs) at the beginning and at the end of a MD run according to the following equation:

$$AP = \frac{SASA_{\text{initial}}}{SASA_{\text{final}}}$$

The solubility of the pentapeptides ($\Delta G_{\text{water-octanol}}$, kcal mol⁻¹) was assessed through the Wimley-White whole residue hydrophobicity scale, which allows dissecting the overall hydrophobicity as a result of the incremental contribution of each single amino acid within a sequence. The self-assembly behavior of the studied combinations was derived from the $\Delta G_{\text{water-octanol}}$ vs. AP plot (Fig. 2). In this chart, the most soluble and less aggregating pentapeptides should be in the lower left part of the reported distribution, while the less soluble and most aggregating ones should be in the upper right part. The reference pentapeptide (DFNKF, red circle in Fig. 2) is located in the left part of the graph, namely it is expected to be quite soluble in water, featuring a medium-high propensity to aggregate.

A limited set of peptide sequences (colored circles in Fig. 2) were selected to validate experimentally the reliability of the CG-MD results, namely prove that the expected self-assembly behaviors are consistent with reality. These peptides were chosen with the intent to be as much as possible representative of the $\Delta G_{\text{water-octanol}}$ vs. AP distribution. A higher number of peptides laying in the central-left region of the graph were selected, since poor water solubility may be a major practical issue, especially in view of potential large-scale preparation and exploitation. We



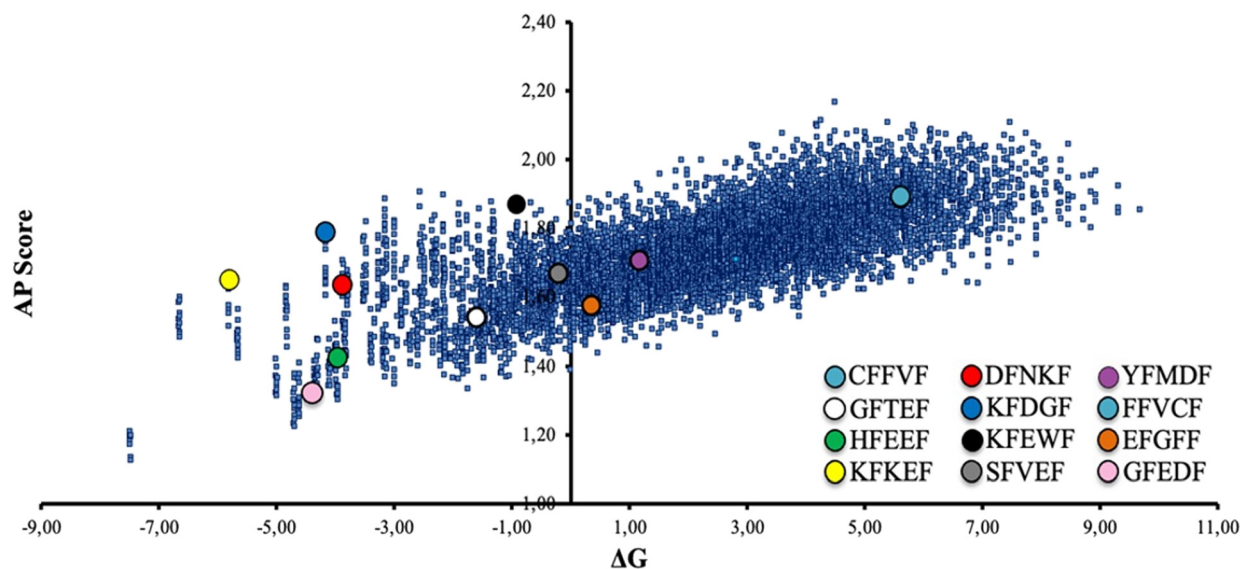


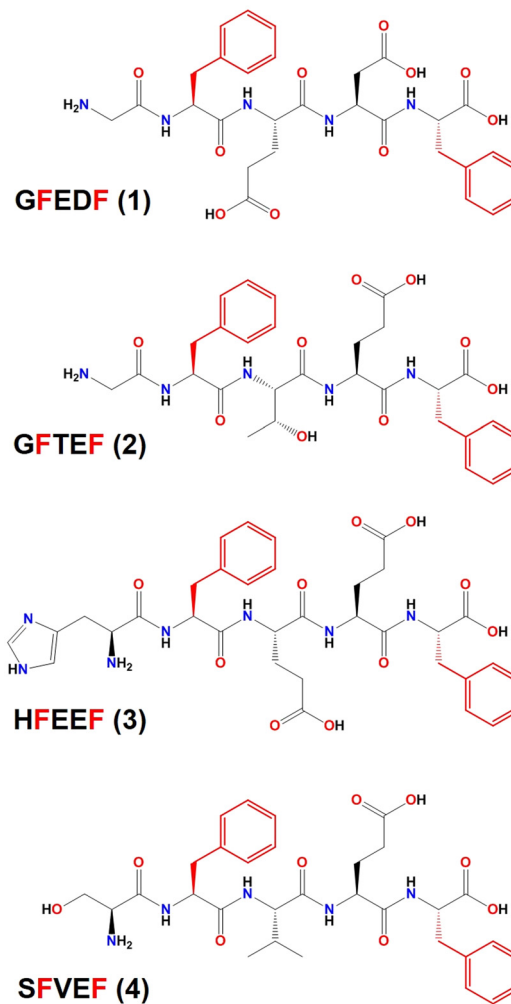
Fig. 2 Graphical representation of solubility ($\Delta G_{\text{water-octanol}}$ kcal mol⁻¹) vs. aggregation propensity (AP) values for the 8000 computed pentapeptides. Colored circles are the sequences that were selected for experimental studies. The reference peptide (DFNKF) is in red.

selected this set of peptides also to assess the role of the N-term amino acid in the self-assembly. Namely, we chose peptides roughly located in the same regions of the graph and differing in the nature (polar, nonpolar, aromatic) of the N-term residue (*e.g.*, GFEDF and HFEEF, located in the same area of the graph and featuring a nonpolar –G– and an aromatic –H– N-term amino acid, respectively).

As the formation of good quality fibrils results from the subtle balance between hydration and aggregation phenomena, it would be oversimplified to think that the best fibril-forming sequences can be found in the upper right region of the plot in Fig. 2, thus experimental confirmation is needed.

The selected sequences were prepared through solid-phase peptide synthesis (SPPS) and characterized for their self-assembly propensity in MilliQ water at the common concentration of 0.4 mM. This value was selected to have a direct comparison among all the pentapeptides, including the ones showing a lower water solubility, namely FFVCF and YFMDF. This finding is coherent with the CG-MD study reported in Fig. 2, as these two sequences are positioned in the right region of the plot.

Of note, four out of twelve of the synthesized peptides (Scheme 1) formed crystals suitable for single crystal X-ray diffraction from the same 0.4 mM solutions within two weeks. Since obtaining good-quality peptide crystals is generally nontrivial, we may argue that the CG-MD applied on the model sequence xFxxF can be considered as a promising tool to statistically forecast not only the self-assembly behaviour, but also the theoretical conditions which may translate into the intrinsic crystallinity of peptide-based systems. In this regard, it may be no coincidence that the crystallized sequences are all located in the same area of the graph in Fig. 2 (lower-left region). This could be an



Scheme 1 Chemical structure of the peptide sequences whose crystal structures have been reported in this work. Phenylalanine residues, whose position was kept in the CG-MD study, are depicted in red.



interesting example of computations applied to biomimetic crystal engineering.

In all the reported crystal structures (Fig. 3 and 4) the peptide strands assemble in β -sheets, which is consistent with the fact that these pentapeptides have been derived from the reference sequence DFNKF, which is well-known for its amyloidogenic propensity. The strand-strand separation, fixed by multiple N-H $\cdots$ O hydrogen bonds (Table S4†) involving the amide moieties of stacked strands, is in the range of 4.7–4.9 Å, in line with the crystal structure of other small peptide fragments previously reported.^{46,47}

Each observed β -sheet and the resulting cross- β spines,⁴⁸ namely the pairing of β -sheets running along the fibril axis, present distinct structural features (Table 1), suggesting that each residue, not only the fixed phenylalanines, has a non-negligible role in driving the self-assembly of the system.

In detail, stacked strands of peptide 1 (GFEDF) give rise to parallel β -sheets, which form the cross- β motif as a result of reciprocal antiparallel pairing between facing β -sheet units (Fig. 3, top). This supramolecular arrangement, with the consequent interdigitation between the side-chains of paired β -sheets, generally dubbed as “steric zipper”,⁴⁹ can be classified as a homosteric zipper of class 1 (parallel, up-up, face-to-face). Specifically, this self-complementarity is strengthened by the occurrence of a strong hydrogen bond involving facing glutamic acid side-chains (OE1A $\cdots$ HE24 distance is 2.15 Å).

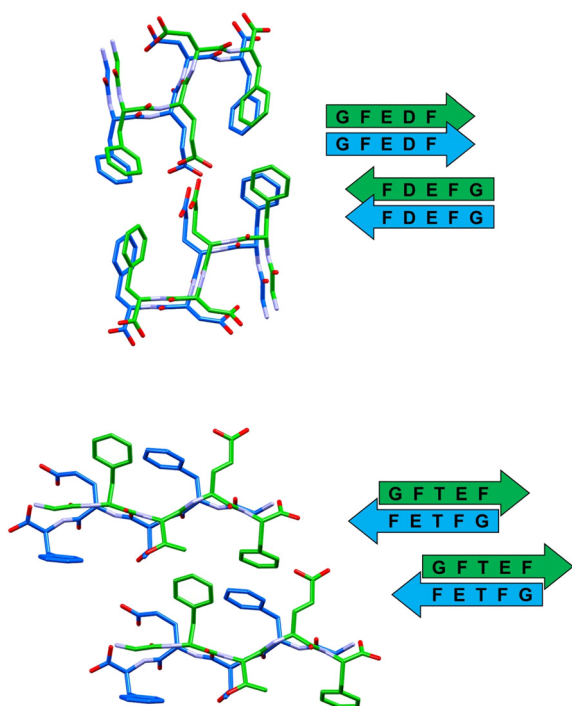


Fig. 3 Stick representation (Mercury 2022.1.0) of crystal structures of peptide 1 (top) and peptide 2 (bottom), showing the essential packing features of the β -sheet arrangements. Hydrogen atoms have been omitted for clarity. Colour code: carbon, green and blue; nitrogen, violet; oxygen, red.

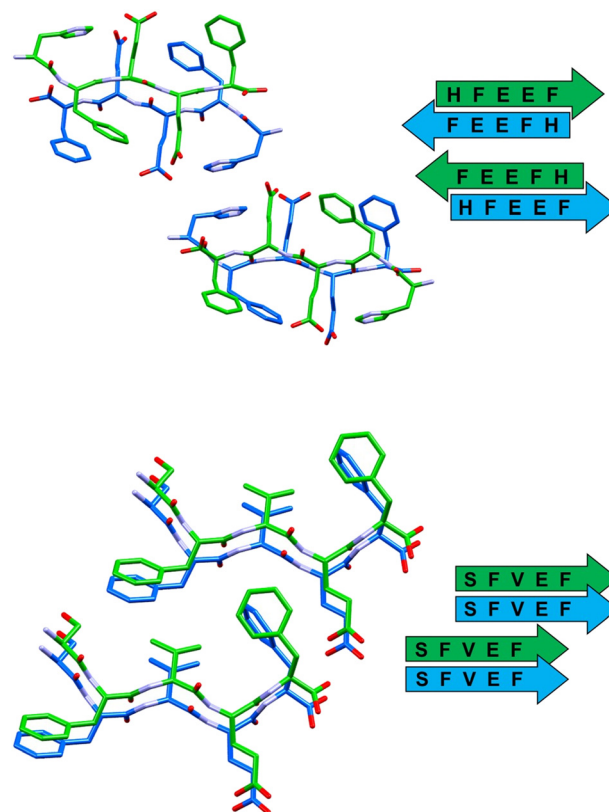


Fig. 4 Stick representation (Mercury 2022.1.0) of crystal structures of peptide 3 (top) and peptide 4 (bottom), showing the essential packing features of the β -sheet arrangements. Hydrogen atoms have been omitted for clarity. Colour code as in Fig. 3.

Peptide 2 (GFTEF) forms antiparallel β -sheets, which pack in a steric zipper of class 6 (antiparallel, up=down, face-to-back). Here, additional stabilization of the zipper arrangement is given by π - π interactions involving the phenylalanine residues of facing β -sheets (Fig. 3, bottom).

The self-assembly of sequence 3 (HFEEF) results in a homosteric zipper of class 5 (antiparallel, up=down, face-to-face), where the most notable interaction is the π - π stacking between imidazole rings of histidine residues (Fig. 4, top).

β -Sheets of peptide 4 (SFVEF) display a parallel orientation (Fig. 4, bottom), forming a steric zipper of class 2 (parallel, up-up, face-to-back), where a strong hydrogen bond between the glutamic acid side chain and the C-terminal of facing sheets (OXT_B5 $\cdots$ HE2_B4, 1.73 Å) aids in keeping the tight interdigitation of the zipper, in addition to other weaker interactions such as C-H $\cdots$ π (involving a -CH₃ group of

Table 1 Structural features of β -sheets and cross- β motifs displayed by peptides 1–4

	β -Sheet orientation	Paired β -sheets	Steric zipper
GFEDF (1)	Parallel	Antiparallel	Class 1
GFTEF (2)	Antiparallel	Parallel	Class 6
HFEEF (3)	Antiparallel	Antiparallel	Class 5
SFVEF (4)	Parallel	Parallel	Class 2



valine or a $-\text{CH}_2-$ of glutamic acid, *i.e.* C-H donors, and the aromatic rings of facing phenylalanines, *i.e.* acceptors).

Overall, these peptide strands adopting an extended conformation interact laterally by means of salt bridges connecting the C-terminal carboxylate and the protonated N-terminal of flanked molecules. This is not the case of **3** (HFEEF), where the C-terminal is not charged and the presence of trifluoroacetate ions functions as a bridge connecting the flanked peptide strands.

The same 0.4 mM water solutions, from which peptides **1–4** were crystallized, were analysed by TEM 48 hours after preparation, in order to visualize the morphology of the supramolecular structures formed by the synthesized peptide sequences. TEM images of the twelve studied peptides display a variety of nanostructures (Fig. 5), whose typology seems to be strongly related to the nature of the amino acid at the N-terminal. Specifically, in the case of charged side-chains (D, aspartic acid; E, glutamic acid; K, lysine) amorphous aggregates (Fig. S2†) or short and poorly defined fibrils (DFNKF, Fig. 5) are observed. Meanwhile, in the case of aromatic or non-polar amino acids (F, phenylalanine; H, histidine; Y, tyrosine; G, glycine) well-developed fibril networks or straight ribbons

can be observed. Once the N-terminal residue is a polar, non-charged amino acid (C, cysteine; S, serine), the resulting morphology may be more connected to other factors, like the nature of the other residues. Indeed, CFFVF forms straight ribbons, while SFVEF (**4**) presents amorphous structures (Fig. S2†). Of note, the observed morphology for peptide **4** seems in contrast with the crystal structure previously described and showing amyloid features (Fig. 4, bottom). In this case the aging time is a key factor in the aggregation kinetics, since TEM images were recorded after 48 hours of aging, while crystals of the same peptide formed in a longer time.

FTIR spectroscopy (Table S2†) of freeze-dried 48 h aged 0.4 mM solutions confirms the amyloid nature of the non-amorphous nanostructures detected through TEM. Here, the amide I region presents sharp bands located around 1630 cm^{-1} that are peculiar of β -sheet structures.⁵⁰ Peptides forming amorphous aggregates show a huge prevalence of signals around 1660 cm^{-1} , consistent with the unstructured material.⁵¹ Interestingly, FTIR reveals that a small β -sheet component, albeit not enough to result in a defined morphology visible by TEM, is even present in most of the ‘amorphous’ peptides. This may indicate that all the studied peptides, in suitable experimental conditions (*e.g.*, higher concentration, varying pH or ionic strength), can potentially self-assemble into fibrils. Considering these findings and the ΔG vs. AP plot (Fig. 2), we may infer that solubility in most cases exerts a stronger impact than AP in the fibrillation propensity of the peptide. Indeed, none of the peptides having lysine at the N-terminal (KFDGF, KFEWF, KFKEF) showed fibril formation neither from 1.5 mM water solutions, that is almost four times the starting concentration that was used for our studies. Thus, peptide solubility should be a crucial factor to consider in view of selecting potential efficient fibril forming sequences from the computations.

In order to assess the formation of a stable supramolecular network, rheological analysis has been conducted to determine eventually the mechanical properties. Shear stress rheological measurements (Fig. S3†) were carried out on the same 0.4 mM solutions (48 h aged) analyzed through TEM. This analysis revealed that most of the peptide solutions behave as liquids, displaying a very scattered signal in the studied time lapse. The only exceptions are FFVCF and YFMDF (the least soluble peptides), which form soft viscoelastic fluid ($G' > G''$) with low elastic modules. Moreover, the strain sweep experiment suggests an improvement of the LVER region for the FFVCF pentapeptide. These results confirm that the hydrophobic nature of amino acids at the N-terminal strongly influences the stability of the fibril networks.

Conclusions

In this contribution we applied a computational method, based on CG-MD, to peptide systems featuring remarkable

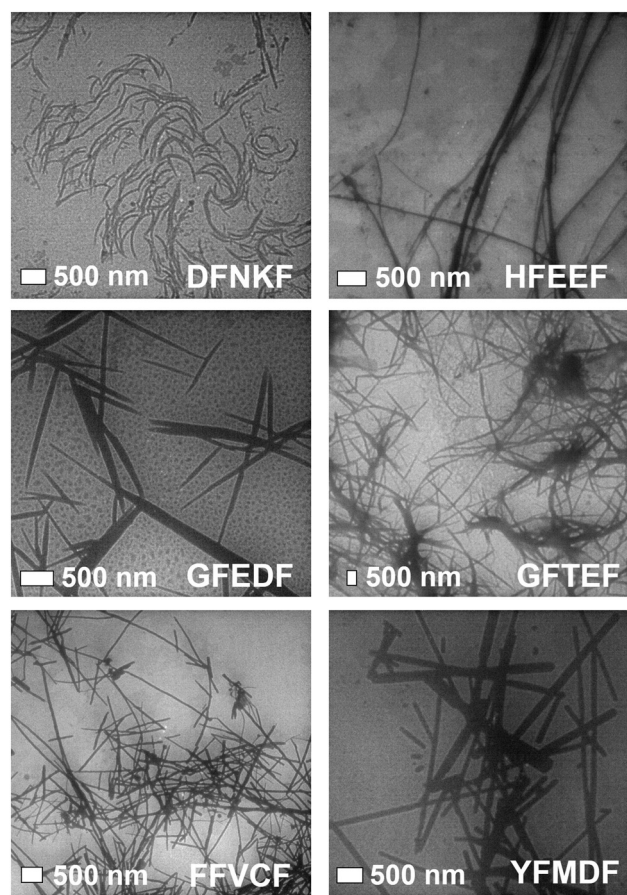


Fig. 5 TEM images of the supramolecular structures formed in water (0.4 mM, aging 48 h) by some of the pentapeptides studied in this work.



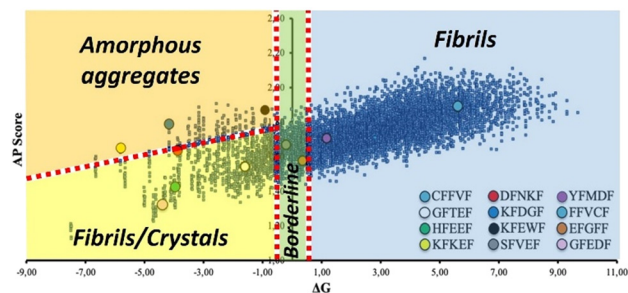


Fig. 6 Representation of the same ΔG vs. AP plot of Fig. 2, where some macro regions associated with specific self-assembly behaviour have been identified based on experimental observations on the synthesized pentapeptides.

structural complexity (five amino acids). Notably, similar methods reported in the literature have been applied on simpler systems (tripeptides). Specifically, starting from a reference sequence (DFNKF) known for its amyloidogenic behavior, we selected a model peptide xFxxF (x = any natural amino acid) where the key residues for fibrillation (F) were kept, and all the possible pentapeptide combinations (20^3) were computed to identify new highly efficient fibril forming sequences. The applied computational model was validated by selecting a limited number of peptide sequences, which were then synthesized and experimentally studied.

TEM and FTIR analyses of the studied peptides confirm the overall reliability of the “fibrillation forecast” resulting from computations (ΔG vs. AP plot), especially considering the many competing/synergic factors playing a role in the self-assembly process of such complex biomimetic systems. In detail, the nature of the residue at the N-terminal was found crucial in the resulting nanostructures formed by these pentapeptides, at least in the tested experimental conditions.

Rheological studies demonstrated that some of these sequences formed nanostructures that confer viscoelastic properties to the water solutions in which they are present. This is a good result in view of the potential employment of these peptides as structuring additive components in materials that need a strengthening of their mechanical performances.

Of note, one third of the synthesized pentapeptides formed good crystals suitable for single crystal X-ray diffraction analysis, which is a remarkable result considering that this kind of system is generally poorly prone to crystallize, especially without the presence of heavy atoms. Importantly, all the crystallized sequences are located in the same sector of the ΔG vs. AP plot (lower-left region, Fig. 6), characterized by medium-high solubility and medium-low AP, which may contain peptides with the tendency to form crystals, possibly thanks to the balance between solubility and low aggregation propensity that allows for slow and ordered (long-range periodicity) self-assembly.

This finding suggests that there should be a reasonable correlation between observed self-assembly behaviors and areas of the ΔG vs. AP graph where certain nanostructures are more likely to be found.

Indeed, we noticed reasonable correlations not only for crystalline assemblies, but also for other observed nanostructures. In detail (Fig. 6), a second area characterized by medium-high solubility and high AP may be populated by peptides that aggregate into amorphous structures, possibly due to the faster aggregation process during solvent evaporation. Finally, a third broad region may feature peptides with medium-to-low solubility and medium AP, which leads to fibril-forming peptides.

Once the calculated ΔG is around zero, subtle variations of the applied experimental conditions may play a major role in the resulting self-assembly (borderline region, Fig. 6). This is the case of EFGFF (amorphous structure) or SFVEF, with the latter forming crystals under slow evaporation conditions or amorphous aggregates under fast evaporation on the TEM grid.

Considering peptide crystallinity, our results may have a huge impact. Indeed, in recent years many efforts have been made in the development peptide-based active pharmaceutical ingredients (APIs), as they guarantee several advantages such as biocompatibility and low production costs.⁵² As crystallinity is usually an important factor in the industrial production and purification of an API, the choice of a reliable CG-model to forecast peptide crystallinity can drastically reduce computational times and costs in the development of peptide-based drugs.

Conflicts of interest

“There are no conflicts to declare”.

Acknowledgements

The authors acknowledge Elettra-Sincrotrone Trieste for providing beamtime under proposal 20185487. P. M. acknowledge the European Research Council (ERC) for Proof-of-Concept grant no. 789815.

Notes and references

- 1 G. M. Whitesides and B. Grzybowski, *Science*, 2002, **295**, 2418–2421.
- 2 S. Scheiner, *Noncovalent Forces*, Springer, 2010, vol. 19.
- 3 C. M. Dobson, *Trends Biochem. Sci.*, 1999, **24**, 329–332.
- 4 P. T. Lansbury and H. A. Lashuel, *Nature*, 2006, **443**, 774–779.
- 5 F. Chiti and C. M. Dobson, *Annu. Rev. Biochem.*, 2006, **75**, 333–366.
- 6 A. Levin, T. A. Hakala, L. Schnaider, G. J. L. Bernardes, E. Gazit and T. P. J. Knowles, *Nat. Rev. Chem.*, 2020, **4**, 615–634.
- 7 A. Lampel, R. V. Ulijn and T. Tuttle, *Chem. Soc. Rev.*, 2018, **47**, 3737–3758.
- 8 P. Besenius, G. Portale, P. H. H. Bomans, H. M. Janssen, A. R. A. Palmans and E. W. Meijer, *Proc. Natl. Acad. Sci. U. S. A.*, 2010, **107**, 17888–17893.
- 9 J. Wang, K. Liu, L. Yan, A. Wang, S. Bai and X. Yan, *ACS Nano*, 2016, **10**, 2138–2143.
- 10 M. A. Sequeira, M. G. Herrera and V. I. Dodero, *Phys. Chem. Chem. Phys.*, 2019, **21**, 11916–11923.



- 11 P. W. J. M. Frederix, G. G. Scott, Y. M. Abul-Haija, D. Kalafatovic, C. G. Pappas, N. Javid, N. T. Hunt, R. V. Ulijn and T. Tuttle, *Nat. Chem.*, 2014, **7**, 30–37.
- 12 A. Pizzi, C. Pigliacelli, G. Bergamaschi, A. Gori and P. Metrangolo, *Coord. Chem. Rev.*, 2020, **411**, 213242.
- 13 A. Marchetti, A. Pizzi, G. Bergamaschi, N. Demitri, U. Stollberg, U. Diederichsen, C. Pigliacelli and P. Metrangolo, *Chem. – Eur. J.*, 2022, **28**, e202104089.
- 14 A. Pizzi, C. Pigliacelli, A. Gori, N. Nonappa, O. Ikkala, N. Demitri, G. Terraneo, V. Castelletto, I. W. Hamley, F. Baldelli Bombelli and P. Metrangolo, *Nanoscale*, 2017, **9**, 9805–9810.
- 15 A. Bertolani, L. Pirrie, L. Stefan, N. Houbenov, J. S. Haataja, L. Catalano, G. Terraneo, G. Giancane, L. Valli, R. Milani, O. Ikkala, G. Resnati and P. Metrangolo, *Nat. Commun.*, 2015, **6**, 7574.
- 16 L. Sori, A. Pizzi, N. Demitri, G. Terraneo, A. Frontera and P. Metrangolo, *CrystEngComm*, 2022, **24**, 7255–7260.
- 17 A. Pizzi, L. Sori, C. Pigliacelli, A. Gautieri, C. Andolina, G. Bergamaschi, A. Gori, P. Panine, A. M. Grande, M. B. Linder, F. Baldelli Bombelli, M. Soncini and P. Metrangolo, *Small*, 2022, **18**, 2200807.
- 18 M. Fayaz-Torshizi and E. A. Müller, *Mol. Syst. Des. Eng.*, 2021, **6**, 594–608.
- 19 A. Kantardjiev, *Soft Matter*, 2021, **17**, 2753–2764.
- 20 K. Shmilovich, R. A. Mansbach, H. Sidky, O. E. Dunne, S. S. Panda, J. D. Tovar and A. L. Ferguson, *J. Phys. Chem. B*, 2020, **124**, 3873–3891.
- 21 A. Jain, C. Globisch, S. Verma and C. Peter, *J. Chem. Theory Comput.*, 2019, **15**, 1453–1462.
- 22 A. Banerjee, C. Y. Lu and M. Dutt, *Phys. Chem. Chem. Phys.*, 2022, **24**, 1553–1568.
- 23 S. Chakraborty, C. M. Berac, M. Urschbach, D. Spitzer, M. Mezger, P. Besenius and T. Speck, *ACS Appl. Polym. Mater.*, 2022, **4**, 822–831.
- 24 A. Lausi, M. Polentarutti, S. Onesti, J. R. Plaisier, E. Busetto, G. Bais, L. Barba, A. Cassetta, G. Campi, D. Lamba, A. Pifferi, S. C. Mande, D. D. Sarma, S. M. Sharma and G. Paolucci, *Eur. Phys. J. Plus*, 2015, **130**, 43.
- 25 W. Kabsch, *Acta Crystallogr., Sect. D: Biol. Crystallogr.*, 2010, **66**, 125–132.
- 26 G. M. Sheldrick, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 3–8.
- 27 P. Emsley, B. Lohkamp, W. G. Scott and K. Cowtan, *Acta Crystallogr., Sect. D: Biol. Crystallogr.*, 2010, **66**, 486–501.
- 28 L. J. Farrugia, *J. Appl. Crystallogr.*, 2012, **45**, 849–854.
- 29 C. F. Macrae, I. Sovago, S. J. Cottrell, P. T. A. Galek, P. McCabe, E. Pidcock, M. Platings, G. P. Shields, J. S. Stevens, M. Towler and P. A. Wood, *J. Appl. Crystallogr.*, 2020, **53**, 226–235.
- 30 L. Schrodinger and W. DeLano, *The PyMOL Molecular Graphics System*, Schrodinger, LLC, 2020.
- 31 *martinize.py*, can be found under <http://md.chem.rug.nl/cgmartini/index.php/downloads/tools/204-martinize>.
- 32 S. Fleming, P. W. J. M. Frederix, I. Ramos Sasselli, N. T. Hunt, R. V. Ulijn and T. Tuttle, *Langmuir*, 2013, **29**, 9510–9515.
- 33 B. Hess, C. Kutzner, D. van der Spoel and E. Lindahl, *J. Chem. Theory Comput.*, 2008, **4**, 435–447.
- 34 S. O. Yesylevskyy, L. V. Schäfer, D. Sengupta and S. J. Marrink, *PLoS Comput. Biol.*, 2010, **6**, 1–17.
- 35 M. McCullagh, T. Prytkova, S. Tonzani, N. D. Winter and G. C. Schatz, *J. Phys. Chem. B*, 2008, **112**, 10388–10398.
- 36 O.-S. Lee, V. Cho and G. C. Schatz, *Nano Lett.*, 2012, **12**, 4907–4913.
- 37 C. A. E. Hauser, R. Deng, A. Mishra, Y. Loo, U. Khoe, F. Zhuang, D. W. Cheong, A. Accardo, M. B. Sullivan, C. Riekel, J. Y. Ying and U. A. Hauser, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 1361–1366.
- 38 C. Wu, H. Lei and Y. Duan, *Biophys. J.*, 2004, **87**, 3000–3009.
- 39 P. W. J. M. Frederix, R. V. Ulijn, N. T. Hunt and T. Tuttle, *J. Phys. Chem. Lett.*, 2011, **2**, 2380–2384.
- 40 C. Guo, Y. Luo, R. Zhou and G. Wei, *ACS Nano*, 2012, **6**, 3907–3918.
- 41 C. Guo, Y. Luo, R. Zhou and G. Wei, *Nanoscale*, 2014, **6**, 2800–2811.
- 42 D. Thirumalai, *Curr. Opin. Struct. Biol.*, 2003, **13**, 146–159.
- 43 L. A. Austin and H. Heath 3rd, *N. Engl. J. Med.*, 1981, **304**, 269–278.
- 44 I. Kedar, M. Ravid and E. Sohar, *Isr. J. Med. Sci.*, 1976, **12**, 1137–1140.
- 45 A. Bertolani, A. Pizzi, L. Pirrie, L. Gazzera, G. Morra, M. Meli, G. Colombo, A. Genoni, G. Cavallo, G. Terraneo and P. Metrangolo, *Chem. – Eur. J.*, 2017, **23**, 2051–2058.
- 46 A. Pizzi, V. Dichiarante, G. Terraneo and P. Metrangolo, *Pept. Sci.*, 2018, **110**, e23088.
- 47 A. Pizzi, L. Catalano, N. Demitri, V. Dichiarante, G. Terraneo and P. Metrangolo, *Pept. Sci.*, 2020, **112**, e24127.
- 48 R. Nelson, M. R. Sawaya, M. Balbirnie, A. Ø. Madsen, C. Riekel, R. Grothe and D. Eisenberg, *Nature*, 2005, **435**, 773–778.
- 49 M. R. Sawaya, S. Sambashivan, R. Nelson, M. I. Ivanova, S. A. Sievers, M. I. Apostol, M. J. Thompson, M. Balbirnie, J. J. W. Wiltzius, H. T. McFarlane, A. Ø. Madsen, C. Riekel and D. Eisenberg, *Nature*, 2007, **447**, 453–457.
- 50 J. Seo, W. Hoffmann, S. Warnke, X. Huang, S. Gewinner, W. Schöllkopf, M. T. Bowers, G. von Helden and K. Pagel, *Nat. Chem.*, 2017, **9**, 39–44.
- 51 Y. Guo and J. Wang, *ChemPhysChem*, 2012, **13**, 3901–3908.
- 52 S. S. Usmani, G. Bedi, J. S. Samuel, S. Singh, S. Kalra, P. Kumar, A. A. Ahuja, M. Sharma, A. Gautam and G. P. S. Raghava, *PLoS One*, 2017, **12**, e0181748.

