

**Measurement of the Triplet Exciton Diffusion Length in  
Organic Semiconductors**

Journal:	<i>Journal of Materials Chemistry C</i>
Manuscript ID	TC-ART-02-2019-000686.R1
Article Type:	Paper
Date Submitted by the Author:	29-Mar-2019
Complete List of Authors:	Rai, Deepesh; University of Minnesota Twin Cities Holmes, Russell; University of Minnesota Twin Cities, Chemical Engineering and Materials Science

## Measurement of the Triplet Exciton Diffusion Length in Organic Semiconductors

Deepesh Rai and Russell J. Holmes\*

Department of Chemical Engineering and Materials Science, University of Minnesota,

Minneapolis, Minnesota 55455, USA

\*Email - rholmes@umn.edu

### Abstract

We present a method to measure the exciton diffusion length ( $L_D$ ) of optically dark triplet excitons in organic semiconductor thin films. In order to directly probe only these states, triplets are optically injected into the material of interest via energy transfer from an adjacent phosphorescent thin film. Injected triplets migrate through the full thickness of the material before undergoing energy transfer to a phosphorescent sensitizer. By measuring photoluminescence from the sensitizer as a function of active layer thickness and sensitizer layer concentration, we are able to extract both  $L_D$  and the transfer rate to the sensitizer. Extraction of the transfer rate is critical, as the assumption of unity quenching can lead to incorrect measurements of  $L_D$ . We validate the method by measuring the singlet exciton diffusion length in the fluorophores tris-(8-hydroxyquinoline)aluminum ( $Alq_3$ ) and 2,3,6,7-tetrahydro-1,1,7,7-tetramethyl-1H,5H,11H-10(2-benzothiazolyl) quinolizine-[9,9a,1gh] coumarin (C545T), and comparing them with values extracted from conventional photoluminescence quenching measurements. The triplet  $L_D$  is subsequently extracted for a series of archetypical fluorescent organic semiconductors with values falling in the range of 15-30 nm. In addition to probing the diffusion of dark triplets, this method also offers the ability

to measure the singlet and triplet  $L_D$  with only a change in injection layer.

## Introduction

Exciton spin plays a central role in the behavior of organic semiconductor thin films and the design of associated devices. Short-lived spin singlet excitons are responsible for fluorescence in organic light-emitting devices (OLEDs) and also form the majority of active materials used for photoconversion in organic photovoltaic cells (OPVs). Long-lived spin triplet excitons are frequently non-radiative, but play important roles as energetic intermediates in OLEDs (as in delayed fluorescence),<sup>1-8</sup> and also as potential active materials in OPVs (via singlet fission).<sup>9-14</sup> In all of these applications, exciton migration plays an important role in the ability to spatially confine or transport energy. For singlets, the direct probing of exciton migration is made possible through photoluminescence measurements.<sup>15,16</sup> For non-radiative triplet excitons, direct photoluminescence-based techniques are not applicable, and alternate methods must be employed.<sup>17-19</sup>

Frequently, device-based measurements are used to indirectly probe triplet diffusion via the byproducts of their dissociation or transfer to a luminescent species. Specifically, these include the fitting of external quantum efficiency spectra in OPVs,<sup>14,20-23</sup> photovoltage measurements,<sup>24-26</sup> photoconductivity measurements,<sup>27</sup> microwave spectroscopy,<sup>28-29</sup> and a variety of measurements in OLED configurations.<sup>30-35</sup> While capable of yielding a value for the exciton diffusion length ( $L_D$ ), intrinsic values of  $L_D$  can only be obtained by carefully accounting for unknown recombination losses and interfacial energy transfer rates.<sup>23</sup> Further, the roles of triplet-triplet annihilation, triplet-polaron quenching, exciton formation zone

migration, and the contribution from singlet diffusion are also difficult to evaluate. Consequently, there is frequently disagreement among different methods of measurement for the triplet  $L_D$ . Non-device-based methods such as delayed fluorescence spread<sup>36-37</sup> and transient absorption spectroscopy<sup>38-39</sup> are capable of probing the intrinsic diffusion length of non-radiative triplets however, their implementation often involves more involved instrumentation. It is for these reasons that the direct probing of non-radiative triplets using a phosphorescent sensitizer<sup>40-41</sup> is of particular interest, due to its ability to distinguish between singlet and triplet transport, as well as prevent bimolecular triplet quenching mechanisms. However, in this method, the quenching efficiency of triplet excitons by the phosphorescent sensitizer is seldom known quantitatively, and hence can cause inaccuracy in reported values. Here, we develop a methodology to resolve this issue and apply the technique to extract the triplet  $L_D$  for a variety of materials.

Figure 1a shows a phosphorescent sensitizer-based methodology for measuring dark triplet excitons in organic semiconductor thin films. Singlet excitons are optically generated in a thin phosphorescent injection layer which undergoes rapid intersystem crossing due to spin-orbit coupling.<sup>42</sup> The resulting triplets are injected directly into the triplet energy level ( $E_T$ ) of an adjacent transport layer (*i.e.* the material of interest), with transfer from the injection layer to the singlet level of the transport layer being frustrated by its endothermicity.<sup>43</sup> Triplet excitons that diffuse through the full thickness of the transport layer may undergo energy transfer to a low-energy phosphorescent sensitizer layer, leading to measurable photoluminescence.<sup>41,44</sup> Exciton diffusion in the transport material is therefore characterized by a transport efficiency

( $\eta_T$ ), defined as the ratio of the exciton collection rate by the sensitizer and the exciton injection rate into the transport layer. Experimentally,  $\eta_T$  is calculated based on the reduction (increase) of photoluminescence (PL) from the injector (sensitizer) in the presence of the transport layer versus a wide energy gap exciton blocking layer ( $\Delta PL^{Injector}$  and  $\Delta PL^{Sensitizer}$ , respectively) as:

$$\eta_T = \frac{\Delta PL^{Sensitizer}}{\Delta PL^{Injector}} \times \frac{\eta_{Injector}^{PL}}{\eta_{Sensitizer}^{PL}} \quad (1)$$

In Eq. (1),  $\eta_{Injector}^{PL}$  and  $\eta_{Sensitizer}^{PL}$  are the outcoupled PL efficiencies of the injector and sensitizer layers. The outcoupled PL efficiencies are determined experimentally as discussed in the Supplementary Information and Fig. S1.

In Eq. (1), the assumption is made that any changes in PL from the injector and sensitizer layers come from the transport of excitons through the material of interest. For this to be accurate, the short-range quenching of excitons in the material of interest by the sensitizer must be efficient. Without knowledge of the quenching efficiency, the extracted  $L_D$  may not reflect the intrinsic behavior of the material. Here, we extract the transfer rate ( $k_Q$ ) from the transport layer to the phosphorescent sensitizer by varying the concentration of the sensitizer in a non-quenching host. In this way, the number of quenching sites (and the quenching efficiency) at the interface is also varied in a predictable manner. Increasing the sensitizer concentration leads to a concomitant increase in  $\eta_T$  until the quenching efficiency reaches unity.

In this work,  $\eta_T$  is measured as function of thickness in order to track the injected triplet exciton density reaching the sensitizer layer. In parallel, measurements of  $\eta_T$  as a function of sensitizer concentration are used to extract  $k_Q$ , and determine an appropriate boundary

condition for the exciton density. A previously published kinetic Monte Carlo (KMC) formalism is used to model the dependence of  $\eta_T$  and extract the triplet diffusion length of the transport layer.<sup>45-46</sup> The injected triplet exciton density in the transport layer is kept small ( $\sim 10^{15} \text{ cm}^{-3}$ ) to avoid triplet-triplet annihilation.<sup>47</sup> The triplet diffusion length is probed in thin films of the archetypical organic semiconductors N,N'-di(1-naphthyl)-N,N'-diphenyl-(1,1'-biphenyl)-4,4'-diamine (NPD), tris-(8-hydroxyquinoline)aluminum (Alq<sub>3</sub>) and bis-(8-hydroxy-2-methylquinoline)-(4-phenylphenoxy)aluminum (BALq). The method is validated against thickness dependent photoluminescence quenching measurements for singlet exciton diffusion in both Alq<sub>3</sub> and 2,3,6,7-tetrahydro-1,1,7,7,-tetramethyl-1H,5H,11H-10(2-benzothiazolyl)quinolizine[9,9a,1gh] coumarin (C545T).

## Experimental

All thin films were deposited using vacuum thermal sublimation (base pressure  $< 7 \times 10^{-7}$  Torr) at a total rate of  $0.1 \text{ nm s}^{-1}$  on glass substrates. Substrates were pre-cleaned by sequentially sonicating in tergitol solution, deionized water, and acetone, followed by rinsing in isopropanol. Material optical constants and film thicknesses were measuring using a J. A. Woollam variable-angle spectroscopic ellipsometer (Cauchy model). Active materials of N, N'-di-1-naphthalenyl-N,N'-diphenyl [1,1':4',1'':4'',1''':4'''-quaterphenyl]-4,4'''-diamine (4P-NPB), tris[2-phenylpyridinato-C<sub>2</sub>,N]iridium(III) (Ir(ppy)<sub>3</sub>), tris(4-carbazoyl-9-ylphenyl)amine (TCTA), 1,3-Bis(N-carbazoyl)benzene (mCP), NPD, C545T and Alq<sub>3</sub> were purchased from Luminescence Technology Corporation; BALq was purchased from Sigma Aldrich, and platinum octaethylporphyrin (PtOEP) and platinum tetra-phenyl-tetra-benzo-porphyrin (PtTPTBP)<sup>48-49</sup> were purchased from Frontier Scientific. The 1,4,5,8,9,11-hexaazatriphenylene

hexacarbonitrile (HATCN)<sup>50</sup> used as a quencher for measuring the singlet  $L_D$  of Alq<sub>3</sub> and C545T was purchased from Luminescence Technology Corporation. In this work, the quenching efficiency of HATCN is assumed to unity, due to its favorable energy offset for electron transfer.<sup>51</sup> Photoluminescence spectra were measured under N<sub>2</sub> purge using a Photon Technology International QuantaMaster 400 Fluorometer equipped with a photomultiplier detection system. Samples were excited at an angle of 70° to sample normal using a monochromatic Xe lamp. The incident light intensity was measured using a S2281 silicon photodiode with a mask of 0.36 cm<sup>2</sup> area. To check for the presence of triplet-triplet annihilation in our measurements, PL spectra were collected at two different pump intensities using a neutral density filter (OD = 0.5). The experimental data was fit using a non-linear least squares methodology where fit error in the calculation of  $L_D$  represents a 95% confidence interval. To detect any crystallinity in the transport layers, X-ray diffraction measurements were carried out at an incident angle of 8 degrees, while samples were scanned using a 2D detector which spans from  $2\theta = 1$  degree to 53 degrees. Triplet energy levels for Ir(ppy)<sub>3</sub> and the sensitizer molecules were determined from room temperature phosphorescence spectra. All other triplet energy levels referenced in this work were determined using the first peak of the phosphorescence taken from literature.<sup>33,52-54</sup>

## Results and Discussion

To demonstrate the validity of the phosphorescent sensitizer-based technique, the method was first applied to measure the singlet diffusion length of Alq<sub>3</sub> and the result compared to well-established thickness-dependent PL quenching-methods.<sup>15</sup> In the generalized scheme of

Fig. 1a, singlet excitons are injected from fluorescent 4P-NPB into Alq<sub>3</sub> by optical pumping at a wavelength of  $\lambda=380$  nm, where absorption occurs mainly in 4P-NPB (Fig. S2a). An injector layer of 4P-NPB is chosen due to its favorable spectral overlap with the absorption spectrum of Alq<sub>3</sub> (see Fig. S2 for active material emission and absorption spectra), enabling efficient Förster energy transfer. Excitons migrate through Alq<sub>3</sub> and are detected by monitoring phosphorescence from a sensitizer layer of PtTPTBP at  $\lambda=770$  nm. The exciton transport efficiency is extracted by comparing changes in PL from PtTPTBP and 4P-NPB (Fig. 2a) in the case when the injector layer of 4P-NPB is present, and where it is replaced by a non-absorbing, wide energy gap spacer layer of mCP. In this way, the increase in photoluminescence from PtTPTBP due to energy transfer from 4P-NPB can be isolated from direct optical excitation (*i.e.* where the injection layer of 4P-NPB is replaced with mCP). The  $\eta_T$  is calculated using Eq. (1) and the associated architecture for measurement is shown in Fig. S1b. The KMC modelling of experimentally measured  $\eta_T$  as a function of Alq<sub>3</sub> thickness results in a single  $L_D$  of  $(5.9 \pm 0.6)$  nm (Fig. 2b).

The architecture for comparing the phosphorescent sensitizer technique against thickness-dependent PL quenching for Alq<sub>3</sub> singlets is shown in Fig. S2b. This method requires the measurement of PL from the material of interest both with and without an adjacent quenching layer, with the ratio of these two quantities fit as a function of thickness for  $L_D$ . Here, the sensitizer layer is replaced by a top quenching layer of HATCN or an exciton blocking layer of mCP for the quenched and unquenched samples, respectively. Singlets are generated in Alq<sub>3</sub> by optical pumping at a wavelength of  $\lambda=440$  nm, where absorption occurs mainly in Alq<sub>3</sub> (Fig.



S2a). A PL ratio is calculated as a function of Alq<sub>3</sub> layer thickness as the ratio of the integrated PL spectrum with and without an adjacent quenching layer (Fig. 2c). The analytical modelling of the experimentally measured PL ratio also yields a singlet L<sub>D</sub> of (5.6 ± 0.4) nm (Fig. 2d),<sup>55</sup> validating the sensitizer-based technique. The method was also verified for the organic fluorescent dye C545T which results in a L<sub>D</sub> of (12.2 ± 0.7) nm as discussed in the Supplementary Information (Fig. S3).

Triplet exciton diffusion is examined for three archetypical organic molecules: NPD, Alq<sub>3</sub> and BAq. Thin films of these materials show no scattering peaks in X-ray diffraction, consistent with previous reports showing that as-grown films of these materials are amorphous.<sup>56-57</sup> All three of these materials are used frequently as transport layers in OLEDs and there is disagreement among the reported L<sub>D</sub> values for both NPD and Alq<sub>3</sub>.<sup>17</sup> Here, the case of NPD is considered first to demonstrate the methodology in detail using the architecture of Fig. 3a. Using a phosphorescent injector layer of Ir(ppy)<sub>3</sub> (E<sub>T</sub> = 2.4 eV),<sup>33</sup> the system is pumped at a wavelength of λ=470 nm, where absorption only occurs in Ir(ppy)<sub>3</sub> and no singlet excitons are generated in NPD (Fig. 3b). Triplets are injected into NPD and those that migrate through the layer may undergo energy transfer to a sensitizer layer of PtOEP (E<sub>T</sub> = 1.9 eV)<sup>33</sup> doped in a wide energy gap host of TCTA (E<sub>T</sub> = 2.8 eV).<sup>52</sup> The η<sub>T</sub> is extracted by comparing the photoluminescence from PtOEP and Ir(ppy)<sub>3</sub> in the case when the transport layer of NPD is present, and where it is replaced by a wide gap spacer layer of TCTA (Fig. 3c). The photoluminescence spectra in Fig. 3c are deconvoluted using the separately measured photoluminescence spectra of PtOEP and Ir(ppy)<sub>3</sub> (Fig. 3b). The decrease in

photoluminescence from Ir(ppy)<sub>3</sub> when adjacent to NPD reflects triplet exciton injection into NPD. Similarly, the corresponding increase in photoluminescence from PtOEP reflects triplet excitons that diffuse through NPD and undergo energy transfer into the sensitizer layer. The boundary condition at the interface between NPD and the sensitizer layer is determined by increasing the concentration of PtOEP molecules from 0.5 wt.% to 30 wt.%. In Fig. 3d, the experimentally measured values of  $\eta_T$  as a function of NPD thickness are similar for PtOEP concentrations >5 wt.%, signifying that all excitons reaching the interface are quenched. The boundary condition is implemented in the KMC model by varying  $k_Q$  to the sensitizer layer with respect to energy transfer ( $k_T$ ) within the layer. The  $k_T$  can be linked to the exciton diffusivity ( $D$ ) and  $L_D$  as:

$$D = \frac{1}{6} \sum_N d^2 k_T(d) = \frac{L_D^2}{\tau} \quad (2)$$

where  $d$  is the intermolecular spacing assuming a cubic lattice,  $\tau$  is the exciton lifetime and  $N$  is the number of molecular hopping sites.<sup>18,58</sup> The KMC modelling of the experimentally measured dependence of  $\eta_T$  on NPD thickness yields  $L_D = (29.8 \pm 1.1)$  nm for the triplet of NPD (Fig. 3b). The measured  $L_D$  is larger than that previously obtained by fitting device photocurrent, likely reflecting device specific recombination losses.<sup>20</sup> The measured  $L_D$  for NPD is smaller than the previous related work which has doped a discontinuous sensitizer layer into the transport layer and assumed that the sensitizer does not significantly perturb the underlying triplet density.<sup>41</sup> The current work suggests that accurate determination of the triplet  $L_D$  depends upon both the sensitizer concentration and boundary condition assumptions. Inaccurate assumptions for a given sensitizer concentration can lead to either overestimation

or underestimation of  $L_D$ . For example, in the current work, the incorrect assumption of perfect quenching by 0.5 wt.% PtOEP would yield an underestimated triplet  $L_D = 12$  nm. Similarly, assuming here that the sensitizer does not perturb the exciton density in the case of quenching by 15 wt.% PtOEP yields an overestimated triplet  $L_D = 52$  nm.

The triplet exciton diffusion lengths of Alq<sub>3</sub> and BAlq are similarly determined using a sensitizer of PtTPTBP ( $E_T = 1.6$  eV) doped in mCP ( $E_T = 2.9$  eV)<sup>53</sup>. The triplets are injected by optically pumping Ir(ppy)<sub>3</sub> at a wavelength of  $\lambda = 470$  nm (Fig. S2a). Triplets diffusing through the transport layer are probed for two different concentration of sensitizer (5 wt.% and 15 wt.% PtTPTBP) in order to check the quenching efficiency of excitons reaching the sensitizer interface. The  $\eta_T$  is calculated using Eq. (1) from the measured photoluminescence of PtTPTBP and Ir(ppy)<sub>3</sub> for the case when the transport layer of BAlq or Alq<sub>3</sub> is present, and when it is replaced by a wide gap spacer layer of mCP. The KMC modelling of the experimentally determined  $\eta_T$  as a function of BAlq and Alq<sub>3</sub> thickness yield  $L_D$  values of  $(16.6 \pm 1.0)$  nm and  $(15.2 \pm 0.9)$  nm, respectively (Fig. 4). The agreement in  $\eta_T$  for two different sensitizer concentrations suggests efficient quenching at the transport layer and sensitizer interface. The measured  $L_D$  for Alq<sub>3</sub> is consistent with the value of  $(14 \pm 9)$  nm previously reported by Baldo et al.<sup>33</sup> using transient analysis of phosphorescence in an OLED.

In addition to offering a means to accurately probe the diffusion of dark triplet excitons, the phosphor sensitized methodology is attractive for its ability to separately probe singlet or triplet exciton transport. Indeed, by simply varying the injecting layer, either the singlet or triplet of the material of interest can be excited. For example, in Alq<sub>3</sub> we find here a singlet

diffusion length of  $L_D = (5.9 \pm 0.6)$  nm and a triplet diffusion length of  $L_D = (15.2 \pm 0.9)$  nm in the same architecture, simply by changing the injection layer from 4P-NPB to Ir(ppy)<sub>3</sub>. Certainly, the Alq<sub>3</sub> triplets have long  $L_D$  as compared to its singlets due to an increased lifetime.<sup>17-19</sup> However, the corresponding increase in  $L_D$  is likely throttled by slow short-range Dexter energy transfer for amorphous organic semiconductor thin films.<sup>59</sup> By coupling measurements of this type with additional time-domain investigations, additional insight can be gained into the factors that impact singlet and triplet migration. Further, the ability to selectively pump the singlet and triplet states should have great utility in decoupling the more complex exciton transport that exists in materials exhibiting singlet fission or TADF.<sup>60</sup> The tri-layer stack used in current work may also be extended to solution-processed active materials. For example, the use of orthogonal solvents is increasingly being demonstrated to prevent re-dissolving in multilayer stacks.<sup>61-66</sup> Further, the combination of a solution-processed bottom layer and vapor-processed top layer would reduce the issue of re-dissolving to only the bottom interface.

## Conclusions

We present a phosphorescent sensitizer-based methodology to accurately extract the exciton diffusion length of non-radiative triplet excitons in organic semiconductor thin films. In order to confirm the validity of the technique, we first extract the  $L_D$  of luminescent singlet excitons in thin films of Alq<sub>3</sub> and C545T, comparing the result to conventional photoluminescence measurements. For measurement of non-radiative triplet excitons, excitons are injected by energy transfer from an adjacent phosphorescent layer. By measuring the

transport efficiency of excitons across the material of interest, triplet  $L_D$  values in the range of 15-30 nm are obtained. The method is also attractive in that values for the singlet and triplet  $L_D$  be extracted for a given material in a common architecture by varying only the exciton injecting material.

### **Conflicts of Interest**

There are no conflicts of interest to declare.

### **Acknowledgements**

This work was supported by National Science Foundation (NSF) Electronics, Photonics and Magnetic Devices under ECCS-1509121. The authors acknowledge C. Clark for X-ray diffraction measurement of as-grown thin films of active materials.

### **Supporting Information**

Methodology to measure outcoupled photoluminescence efficiency ratio for the architectures probed in this study. Absorption coefficient and normalized photoluminescence spectrum of the materials probed in this study. Architectures for extracting the  $\eta_T$  as a function of transport layer thickness.

**References:**

1. H. Uoyama, K. Goushi, K. Shizu, H. Nomura and C. Adachi, *Nature*, 2012, **492**, 234–238.
2. D. Di, A.S. Romanov, L. Yang, J. M. Richter, J. P. H. Rivett, S. Jones, T. H. Thomas, M. A. Jalebi, R. H. Friend, M. Linnolahti, M. Bochmann, D.M. Credginton, *Science*, 2017, **356**(6334) 159-163.
3. N. A. Kukhta, T. Matulaitis, D. Volyniuk, K. Ivaniuk, P. Turyk, P. Stakhira, J. V. Grazulevicius, and A. P. Monkman, *J. Phys. Chem. Lett.*, 2017, **8**, 6199–6205.
4. Z. Yang, Z. Mao, Z. Xie, Y. Zhang, S. Liu, J. Zhao, J. Xu, Z. Chi, and M. P. Aldred, *Chem. Soc. Rev.*, 2017, **46**, 915.
5. T. Hosokai, H. Matsuzaki, H. Nakanotani, K. Tokumaru, T. Tsutsui, A. Furube, K. Nasu, H. Nomura, M. Yahiro, C. Adachi, *Sci. Adv.*, 2017, **3**, 5.
6. T. J. Penfold, F. B. Dias and A. P. Monkman, *Chem. Commun.*, 2018, **54**, 3926.
7. M. Y. Wong, and E. Z. Colman, *Adv. Mater.*, 2017, **29**, 1605444.
8. D. Di, L. Yang, J. M. Richter, L. Meraldi, R. M. Altamimi, A.Y. Alyamani, D. Credginton, K. P. Musselman, J. L. MacManus-Driscoll, and R. H. Friend, *Adv. Mater.*, 2017, **29**, 1605987
9. A. Rao, and R. H. Friend, *Nat. Rev. Mater.*, 2017, **2**, 17063.
10. D. N. Congreve, J. Lee, N. J. Thompson, E. Hontz, S. R. Yost, P. D. Reuswig, M. E. Bahlke, S. Reineke, T. V. Voorhis, M. A. Baldo, *Science*, 2013, **340**(6130), 334-337.
11. A. Rao, M. W. B. Wilson, J. M. Hodgkiss, S. Albert-Seifried, H. Bässler and R. H. Friend, *J. Am. Chem. Soc.*, 2010, **132** (36), 12698–12703.

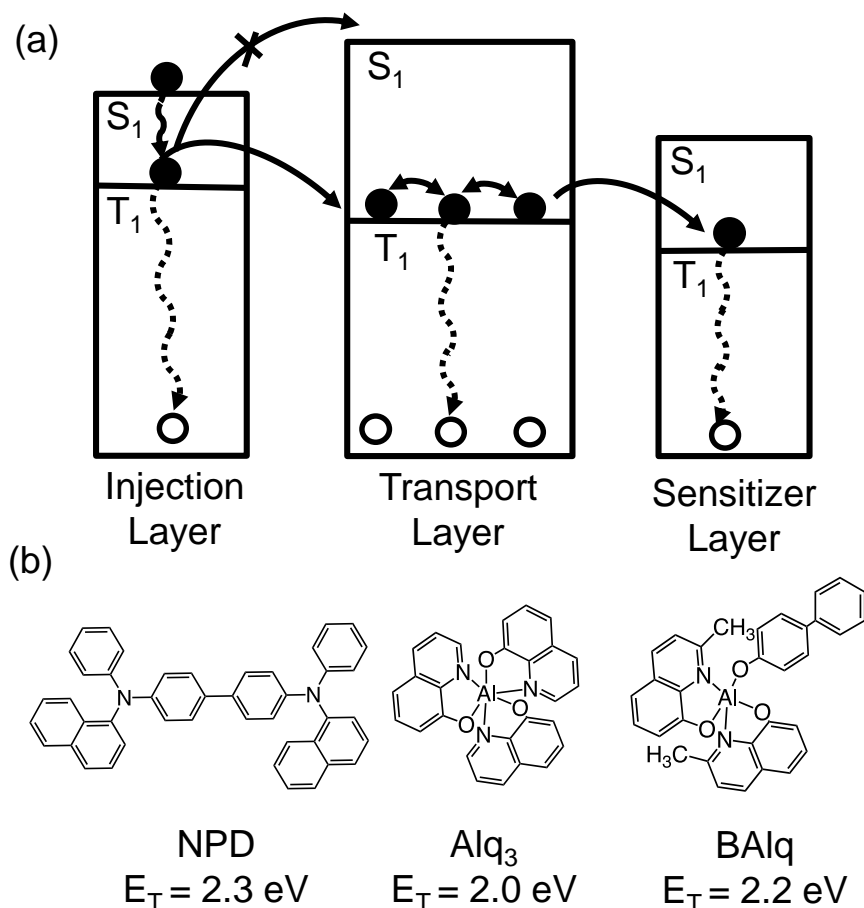
12. J. Lee, P. Jadhav, P. D. Reuswig, S. R. Yost, N.J. Thompson, D. N. Congreve, E. Hontz, T. V. Voorhis, and M. A. Baldo, *Chem. Res.*, 2013, **46** (6), 1300–1311.
13. M. B. Smith, J. Michl, *Annu. Rev. Phys. Chem.*, 2013, **64**, 361–86.
14. M. Tabachnyk, B. Ehrler, S. Bayliss, R. H. Friend and N. C. Greenham, *Appl. Phys. Lett.*, 2013, **103**, 153302.
15. P. Peumans, A. Yakimov, and S. R. Forrest, *J. Appl. Phys.*, 2003, **93**, 3693.
16. R. R. Lunt, N. C. Giebink, A. A. Belak, J. B. Benziger, and S. R. Forrest, *J. Appl. Phys.*, 2009, **105**, 053711.
17. O. V. Mikhnenko, P. W. M. Blom, T.-Q. Nguyen, *Energy Environ. Sci.*, 2015, **8**, 1867–1888.
18. S. M. Menke and R. J. Holmes, *Energy Environ. Sci.*, 2014, **7**, 499-512.
19. G. J. Hedley, A. Ruseckas, and I. D. W. Samuel, *Chem. Rev.*, 2017, **117** (2), 796-837.
20. W. A. Luhman, and R. J. Holmes, *Appl. Phys. Lett.*, 2009, **94**, 153304.
21. B. Siegmund, M. T. Sajjad, J. Widmer, D. Ray, C. Koerner, M. Riede, K. Leo, I. D. W. Samuel, and K. Vandewal, *Adv. Mater.*, 2017, **29**, 1604424.
22. B. P. Rand, S. Schols, D. Cheyns, H. Gommans, C. Girotto, J. Genoe, P. Heremans, J. Poortmans, *Organic Electronics*, 2009, **10** (5), 1015-1019.
23. T. Zhang, D. B. Dement, V. E. Ferry and R. J. Holmes, 2019, *Nat. Commun.*, 10, 1156
24. T. K. Mullenbach, I. J. Curtin, T. Zhang, and R. J. Holmes, *Nat. Commun.*, 2017, **8**, 14215.
25. I. J. Curtin and R. J. Holmes, *Adv. Energy Mater.*, 2018, **8** (13), 1702339.
26. T. Zhang, and R. J. Holmes, *J. Mater. Chem. C*, 2017, **5** (45), 11885-11891.
27. H. Najafov, B. Lee, Q. Zhou, L. C. Feldman, V. Podzorov, *Nat. Mater.*, 2010, **9**, 938–943.
28. J. E. Kroeze, T. J. Savenije, L. P. Candeias, J. M. Warman and L. D. A. Siebbeles, *Sol.*

- Energy Mater. Sol. Cells*, 2005, **85**, 189–203.
29. T. J. Savenije, A. J. Ferguson, N. Kopidakis, and G. Rumbles, *J. Phys. Chem. C*, 2013, **117**, 46, 24085-24103.
30. J. Wünsche, S. Reineke, B. Lüssem, and K. Leo, *Phys. Rev. B*, 2010, **81**, 245201.
31. M. A. Baldo, D. F. O'Brien, M. E. Thompson, S. R. Forrest, *Phys. Rev. B*, 1999, **60**, 14422.
32. Y. Luo and H. Aziz, *J. Appl. Phys.*, 2010, **107**, 094510.
33. M. A. Baldo and S. R. Forrest, *Phys. Rev. B*, 2000, **62**, 10958.
34. M. Lebental, H. Choukri, S. Chenais, S. Forget, A. Siove, B. Geffroy, and E. Tutis, *Phys. Rev. B*, 2009, **79**, 165318.
35. Y. C. Zhou, L. L. Ma, J. Zhou, X. M. Ding, and X. Y. Hou, *Phys. Rev. B*, 2007, **75**, 132202.
36. G. Akselrod, P. Deotare, N. Thompson, J. Le, W. Tisdale, M. Baldo, V. Menon, V. Bulović, *Nat. Commun.*, 2014, **5**.
37. P. Irkhin and I. Biaggio, *Phys. Rev. Lett.*, 2011, **107**, 017402.
38. A. D. Poletayev, J. Clark, M. W. B. Wilson, A. Rao, Y. Makino, S. Hotta and R. H. Friend, *Adv. Mater.*, 2014, **26**, 919–924.
39. C. Grieco, G. S. Doucette, R. D. Pensack, M. M. Payne, A. Rimshaw, G. D. Scholes, J. E. Anthony, and J. B. Asbury, *J. Am. Chem. Soc.*, 2016, **138**, 16069–16080.
40. N. C. Giebink, Y. Sun and S. R. Forrest, *Org. Electron.*, 2006, **7**, 375–386.
41. O. V. Mikhnenko, R. Ruiter, P. W. M. Blom, and M. A. Loi, *Phys. Rev. Lett.*, 2012, **108**, 137401.
42. M. A. Baldo, D. F. O'Brien, Y. You, A. Shoustikov, S. Sibley, M. E. Thompson, and S. R.

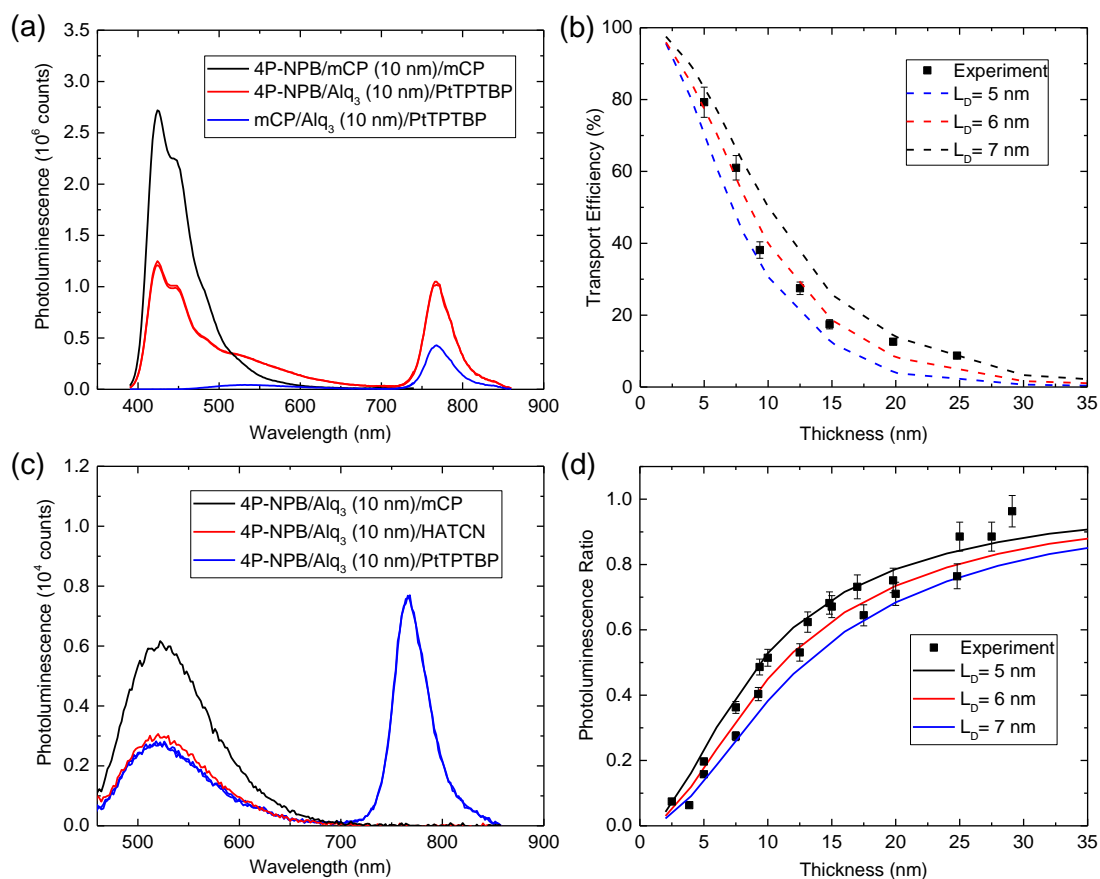


- Forrest, *Nature*, 1999, **395**, 151-154.
43. M. Wu, D. N. Congreve, M. W. B. Wilson, J. Jean, N. Geva, M. Welborn, T. V. Voorhis, V. Bulović, M. G. Bawendi & M. A. Baldo, *Nat. Photonics*, 2016, **10**, 31–34.
44. N. J. Thompson, M. W. B. Wilson, D. N. Congreve, P. R. Brown, J. M. Scherer, T. S. Bischo, M. Wu, N. Geva, M. Welborn, T. V. Voorhis, V. Bulovic, M. G. Bawendi, and M. A. Baldo, *Nat. Mater.*, 2014, **13**, 1039–1043.
45. S. M. Menke, T. K. Mullenbach, R. J. Holmes, *ACS Nano.*, 2015, **9**, 4543–4552.
46. D. Rai, and R. J. Holmes, *Phys. Rev. Appl.*, 2019, **11**, 014048.
47. B. W. Caplins, T. K. Mullenbach, R. J. Holmes, and D.A. Blank, *J. Phys. Chem. C*, 2015, **119** (49), 27340–27347.
48. K. W. Hershey, J. Suddard-Bangsund, G. Qian, R. J. Holmes, *Appl. Phys. Lett.*, 2017, **111**, 113301.
49. N. C. Erickson, R. J. Holmes, *Adv. Funct. Mater.*, 2013, **23**, 5190-5198.
50. S. M. Menke and R. J. Holmes, *ACS Appl. Mater. Interfaces*, 2015, **7** (4), 2912-2918.
51. Y. Kim, J. W. Kim, and Y. Park, *Appl. Phys. Lett.*, 2009, **94**, 063305.
52. T. Kato, T. Mori, T. Mizutani, *Thin Solid Films*, 2001, **393**, 109.
53. R. J. Holmes and S. R. Forrest, *Appl. Phys. Lett.*, 2003, **82**, 2422.
54. S. Tokito and I. Tanaka, *Electrochemistry*, 2008, **76**, 24-31.
55. L. A. A. Pettersson, L. S. Roman, and O. Inganäs, *J. Appl. Phys.*, 1999, **86**, 487–496.
56. M. Brinkmann, G. Gadret, M. Muccini, C. Taliani, N. Masciocchi, and A. Sironi, *J. Am. Chem. Soc.*, 2000, **122** (21), 5147–5157.

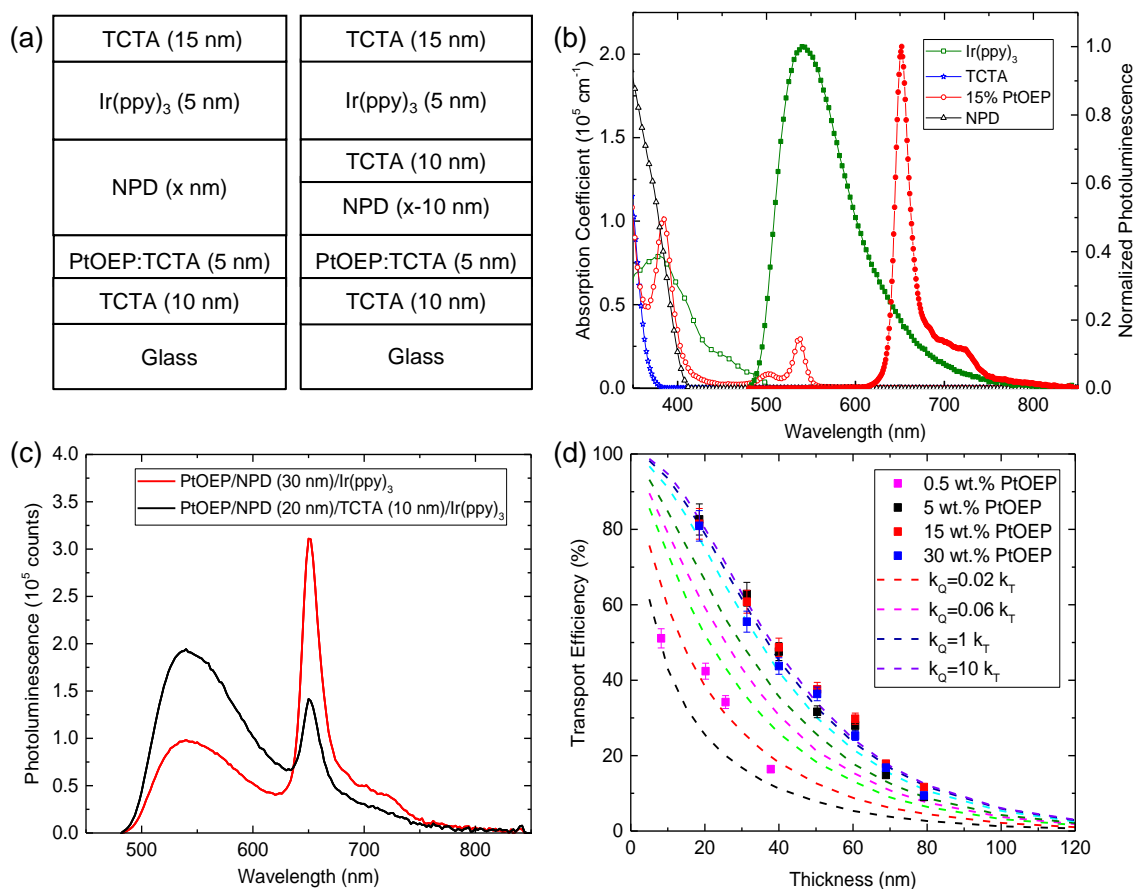
57. J. Ding, Q. Wang, L. Zhao, D. Ma, L. Wang, X. Jinga and F. Wanga, *J. Mater. Chem.*, 2010, **20**, 8126-8133.
58. R. C. Powell and Z. G. Soos, *J. Lumin.*, 1975, **11**, 1.
59. S. R. Yost, E. Hontz, S. Yeganeh, and T. V. Voorhis, *J. Phys. Chem. C*, 2012, **116**, 17369–17377.
60. S. M. Menke and R. J. Holmes, *J. Phys. Chem. C*, 2016, **120**, 16, 8502-8508.
61. B. E. Lassiter, J. D. Zimmerman, and S. R. Forrest, *Appl. Phys. Lett.*, 2013, **103**, 123305.
62. N. Aizawa, Y. J. Pu, M. Watanabe, T. Chiba, K. Ideta, N. Toyota, M. Igarashi, Y. Suzuri, H. Sasabe, and J. Kido, *Nat. Commun.*, 2014, **5**, 5756.
63. Z. Hu, F. Huang, Y. Cao, *Small Methods*, 2017, **1**, 1700264.
64. S. Wang, B. Zhang, Y. Wang, J. Ding, Z. Xie and L. Wang, *Chem. Commun.*, 2017, **53**, 5128-5131.
65. T. L. Murrey, K. Guo, J. T. Mulvey, O. A. Lee, C. Cendra, Z. I. Bedolla-Valdez, A. Salleo, J. F. Moulin, K. Hong, and A. J. Moulé, *J. Mater. Chem. C*, 2019, **7**, 953-960.
66. Q. Dao, A. Fujii, and M. Ozaki, *Jpn. J. Appl. Phys.*, 2016, **55**, 03DB01.



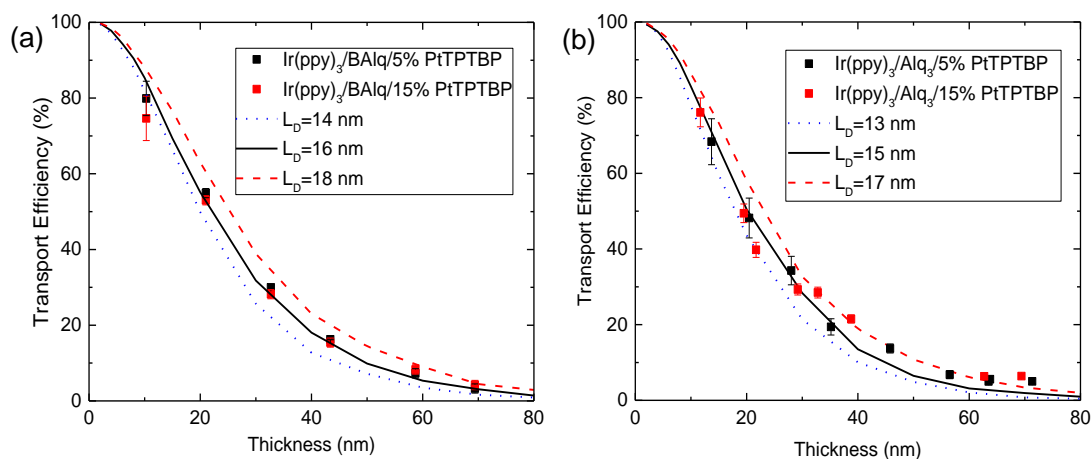
**Figure 1:** (a) A generalized scheme for probing the diffusion length of non-radiative triplet excitons in organic semiconductor thin films. Excitons are injected into the transport layer by energy transfer from an adjacent phosphorescent injection layer. Excitons diffuse through the material of interest (transport layer) before undergoing energy transfer to the phosphorescent sensitizer layer. The triplet energy levels of the three layers are selected to ensure downhill energy transfer from the injection layer to the sensitizer layer. (b) Molecular structure and triplet energy ( $E_T$ ) of three archetypical fluorescent semiconductors investigated in this study.



**Figure 2:** (a) Photoluminescence spectra collected for the multilayer structures (4P-NPB/mCP ( $x$  nm)/mCP or 4P-NPB/Alq<sub>3</sub> ( $x$  nm)/10 wt.% PtTPTBP:mCP and mCP/Alq<sub>3</sub> ( $x$  nm)/10 wt.% PtTPTBP:mCP) used to extract the singlet diffusion length of Alq<sub>3</sub> using the phosphorescent sensitizer-based approach. The structure is pumped at a wavelength of  $\lambda=380$  nm where a majority of excitons are generated in the 4P-NPB injection layer. The excitons that diffuse through Alq<sub>3</sub> are detected using 10 wt.% PtTPTBP doped in mCP. (b) Experimental and simulated (lines) transport efficiency as a function of transport layer thickness. The  $L_D$  is extracted from a KMC simulation. (c) Photoluminescence spectra used to extract the singlet diffusion length of Alq<sub>3</sub> using thickness dependent PL quenching. The structure (4P-NPB/Alq<sub>3</sub> ( $x$  nm)/mCP and 4P-NPB/Alq<sub>3</sub> ( $x$  nm)/HATCN) is pumped at a wavelength of  $\lambda=440$  nm where all the excitons are generated in Alq<sub>3</sub>. (d) Photoluminescence ratio versus thickness for determination of the singlet exciton diffusion length of Alq<sub>3</sub>.

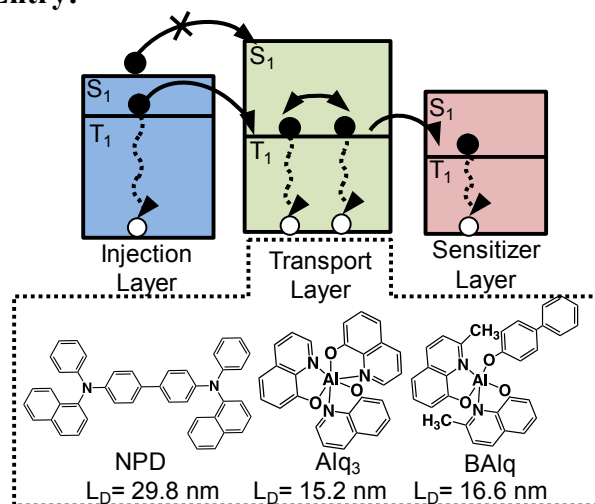


**Figure 3:** (a) Architectures used to extract the transport efficiency and triplet  $L_D$  for NPD as a function of thickness. The structure is pumped at a wavelength of  $\lambda=470$  nm where the majority of excitons are generated in Ir(ppy)<sub>3</sub> injection layer. (b) Absorption coefficient (open symbols) and normalized photoluminescence (solid symbols) for the materials of interest. Spectra for PtOEP are collected for films of 15 wt.% PtOEP in TCTA. (c) Representative photoluminescence spectra for multilayer structures (15 wt.% PtOEP:TCTA/NPD (30 nm)/Ir(ppy)<sub>3</sub> and 15 wt.% PtOEP:TCTA/NPD (20 nm)/TCTA (10 nm)/Ir(ppy)<sub>3</sub>) used to probe the triplet diffusion length of NPD using the phosphorescent sensitizer-based approach. (d) Experimental (symbols) and simulated (line) transport efficiency as a function of transport layer thickness. The  $\eta_T$  versus thickness is plotted for various values of transfer rate ( $k_Q$ ) to the sensitizer relative to bulk energy transfer ( $k_T$ ) for a fixed NPD triplet  $L_D=30$  nm.



**Figure 4:** Experimental (symbols) and simulated (lines) transport efficiency as a function of BAQ (a) and Alq<sub>3</sub> (b) thickness for different values of the triplet L<sub>D</sub>. Multiple sensitizer concentrations are considered to ensure unity quenching at transport layer and sensitizer interface.

## Table of Contents Entry:



We develop a methodology to measure the diffusion of dark triplet excitons in organic semiconductor thin films using a phosphorescent sensitizer-based approach that explicitly quantifies quenching efficiency by varying sensitizer concentration.