



Aggregation-Free Sensitizer Dispersion in Rigid Ionic Crystals for Efficient Solid-State Photon Upconversion and Demonstration of Defect Effects

Journal:	<i>Journal of Materials Chemistry C</i>
Manuscript ID	TC-ART-02-2018-000977.R1
Article Type:	Paper
Date Submitted by the Author:	28-Mar-2018
Complete List of Authors:	Ogawa, Taku; Kyushu University, Chemistry and Biochemistry Yanai, Nobuhiro; Kyushu University, Chemistry and Biochemistry Fujiwara, Saiya; Kyushu University, Chemistry and Biochemistry Nguyen, Thuc-Quyen; University of California, Santa Barbara, Chemistry and Biochemistry Kimizuka, Nobuo; Kyushu University, Chemistry and Biochemistry



Aggregation-Free Sensitizer Dispersion in Rigid Ionic Crystals for Efficient Solid-State Photon Upconversion and Demonstration of Defect Effects

Received 00th January 20xx,
Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

www.rsc.org/

Taku Ogawa,^a Nobuhiro Yanai,^{*a,b} Saiya Fujiwara,^a Thuc-Quyen Nguyen,^c and Nobuo Kimizuka^{*a}

Solid-state photon upconversion based on triplet-triplet annihilation (TTA-UC) has attracted much interest because of its potential to circumvent the loss of sub-bandgap photons in photovoltaic cells. There are two important long-standing questions for TTA-UC in solid crystals. Why is the UC efficiency often low in crystalline systems? What is the rational strategy to construct efficient upconverting crystals? In this work, these issues are addressed by employing a simple model system where ionic interactions play a key role. When crystals of an anthracene-based ionic acceptor (emitter) are grown in the presence of anionic donor (sensitizer) molecules, the donor molecules are spontaneously taken up and dispersed homogeneously in acceptor crystals without aggregation. Highly efficient UC is achieved as a consequence of quantitative triplet energy transfer (TET) from the incorporated donor to the surrounding acceptor. The fact is found that the mechanical grinding of the donor-doped single crystals lead to a significant decrease in UC efficiency, suggesting that trap sites formed in the crystals have a significant negative impact on the UC performance. The important fundamental knowledge obtained from the current ionic crystal system offers rational design guidelines towards the development of efficient TTA-UC systems in the solid-state.

Introduction

In recent decades, solar energy conversion devices such as photovoltaic cells have played the major role in the clean energy production.¹ While the silicon-based technology has realized the highly efficient and low-cost photovoltaic cells, there is a well-known efficiency upper limit of 34% for the single-junction photovoltaic cells under the AM 1.5 G condition, so-called Shockley-Queisser limit.² This theoretical limit comes from the wavelength mismatch between the solar irradiance spectrum and device absorption. Therefore, if sub-bandgap photons can be recovered, it is possible to go beyond the Shockley-Queisser limit and it should contribute the resolution of global energy issues. One of the promising solutions to circumvent such wavelength mismatch is photon upconversion (UC). UC is an energy upshifting methodology that can convert longer-wavelength light (lower energy photons) into shorter-wavelength light (higher energy photons). Among several UC mechanisms such as two-photon absorption (TPA)^{3,4} or energy

transfer-based UC in lanthanide-based nanocrystals,^{5,6} triplet-triplet annihilation (TTA)-based UC has particularly attracted attention due to its occurrence at much lower excitation intensity compared with other mechanisms.⁷⁻¹⁹ In the typical TTA-UC scheme, donor (sensitizer) molecules absorb light and undergo intersystem crossing (ISC) to become triplet excited state, which is followed by triplet energy transfer (TET) to acceptor (emitter) molecules. The acceptor triplets (T_1) collide to show TTA, and resulting acceptor excited singlets (S_1) exhibit upconverted delayed fluorescence.

Although high UC efficiencies of around 30% have been achieved in solution because of the ease of molecular diffusion and collision processes,⁷⁻⁹ it is indispensable to develop solid upconversion materials for the real-world application of TTA-UC towards solar energy utilization devices such as photovoltaic cells. Several approaches have been proposed including droplets in rigid matrices,^{12,20} molecular diffusion in rubbery polymers,²¹⁻²⁶ and triplet energy migration (TEM) in dense chromophore assemblies.²⁷⁻³⁴ Among them, TEM-based UC in crystalline materials has the potential to attain an ultimate UC system with high UC efficiency at low excitation intensity thanks to fast TEM in ordered chromophore arrays. However, most of the crystalline TEM-UC systems have suffered from the phase separation of donor molecules in acceptor crystals, resulting in low TET efficiency.^{10,28} In recent years, a few strategies have been reported to overcome this problem. For example, the modification of acceptor units with flexible alkyl chains can create the room for donor molecules to be accommodated in acceptor crystals.³¹ As another approach, fast and kinetically-controlled crystal growth enables to trap donor molecules in

^a Department of Chemistry and Biochemistry, Graduate School of Engineering, Center for Molecular Systems (CMS), Kyushu University, 744 Moto-oka, Nishi-ku, Fukuoka 819-0395, Japan. *E-mail: yanai@mail.cstm.kyushu-u.ac.jp, n-kimi@mail.cstm.kyushu-u.ac.jp.

^b PRESTO, JST, Honcho 4-1-8, Kawaguchi, Saitama 332-0012, Japan.

^c Center for Polymers and Organic Solids, Department of Chemistry and Biochemistry, University of California at Santa Barbara, Santa Barbara, CA 93106, USA.

Electronic Supplementary Information (ESI) available: absorption and emission spectra, IR spectra, PXRD patterns, excitation intensity dependence of UC emission intensity, and UC emission decays. See DOI: 10.1039/x0xx00000x

ARTICLE

rigid acceptor crystals during the crystallization process.³⁴ In these approaches, the interactions operating among donor and acceptor molecules are weak van der Waals dispersion forces. However, these dispersion-force based strategies unfortunately sacrificed the advantages of crystalline systems; they reduced crystal regularity in exchange for the homogeneous donor accommodation, which inhibited the fast and efficient energy migration. In addition, while defects caused by disordered structures were suspected to act as quenching sites for excitons,³⁵⁻³⁷ there have been no reports to directly prove such situation in TTA-UC. Thus, it remains a grand challenge to solve these issues and to develop highly efficient solid UC systems. To find a clue of the relationship between the crystal quality and photophysical properties involved in UC, it is desired to introduce specific interactions to improve the structural integrity of the mixed crystals and to develop a rational strategy that simultaneously fulfills the controlled molecular dispersion of donors in acceptor crystals and the maintenance of regularity in the whole crystalline systems.

Here we show that an introduction of ionic interactions as additional cohesive interactions can suppress aggregation of donor molecules in acceptor ionic crystals without losing the high crystal regularity. The ionic interactions can compensate the inherent structural mismatch between the donor and acceptor molecules since they exert the major interaction in the crystal formation process. As a proof-of-concept, one of the simplest anionic acceptors 9,10-anthracenedicarboxylate (ADC) was employed as a model system, and it was combined with dicyclohexyl ammonium (DCA) cations to form ionic crystals (DCA)₂ADC (Fig. 1a). When the ionic crystals were prepared in the presence of an anionic donor, palladium mesoporphyrin (PdMesOP), the donor molecules were successfully introduced into the crystals. Remarkably, the accommodated donor molecules were found to be molecularly dispersed, resulting in almost 100% donor-to-acceptor TET (Fig. 1b, right). The ionic interactions play a key role in this excellent donor dispersion, as evidenced by the aggregation of commonly-used nonionic donor Pt(II) octaethylporphyrin (PtOEP) in the same ionic crystals (Fig. 1b, left). The impact of crystal defects upon TTA-UC properties was demonstrated for the first time by comparing UC efficiency and emission decay profiles for single crystals and mechanically-ground powder samples. This work offers an unequivocal answer to the long-standing questions; what makes the UC efficiency in crystalline systems low, and how to rationally achieve efficient UC in solid crystals. While the main objective of this work is to prove the concept in the simple model system, the generalization of obtained fundamental knowledge would open a path towards the realization of ultimate solid-state upconverters.

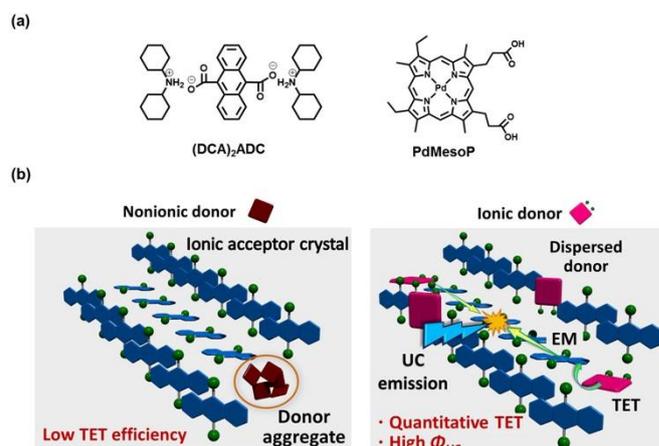


Figure 1. (a) Chemical structures of (DCA)₂ADC and PdMesOP. (b) Schematic illustration of the concept of this study. Nonionic donor molecules aggregate in the acceptor ionic crystals, which ends up with poor TET efficiency (left). On the other hand, ionic donor molecules are molecularly dispersed in the acceptor ionic crystals, resulting in high TET and UC efficiency (right).

Experimental section

General methods

All chemicals were used as received otherwise noted. H₂ADC, dicyclohexylamine and PtOEP were purchased from Aldrich. PdMesOP was purchased from Frontier Scientific. 9,10-diphenylanthracene (DPA) was purchased from TCI and purified by sublimation.

UV-visible absorption spectra were recorded on a JASCO V-670 spectrophotometer. Fluorescence spectra were measured by using a PerkinElmer LS 55 fluorescence spectrometer. Single crystal X-ray data were collected on a CCD diffractometer (Rigaku Saturn VariMax) with graphite-monochromated Mo K α radiation (λ_{ex} = 0.71070 Å). Powder X-ray diffraction (PXRD) analyses were conducted on a BRUKER D2 PHASER with a Cu K α source (λ_{ex} = 1.5418 Å). Scanning electron microscope (SEM) images were obtained by using a Hitachi S-5000.

For TTA-UC measurements, the samples were sealed between quartz plates by using hot-melt adhesive in an Ar-filled glove box ([O₂] < 0.1 ppm). For TTA-UC emission spectra, a diode laser (532 nm, 200 mW, RGB Photonics) was used as an excitation source. The laser power was controlled by combining a software (Ltune) and a variable neutral density filter and measured using a PD300-UV photodiode sensor (OPHIR Photonics). The laser beam was focused on a sample using a lens. The diameter of the laser beam ($1/e^2$) was measured at the sample position using a CCD beam profiler SP620 (OPHIR Photonics). A typical area of laser irradiation spot estimated from the diameter was 2.9×10^{-4} cm². The emitted light was collimated by an achromatic lens, the excitation light was removed using a notch filter (532 nm), and the emitted light was again focused by an achromatic lens to an optical fibre connected to a multichannel detector MCPD-9800 (Otsuka Electronics). Time-resolved photoluminescence lifetime measurements were carried out by using a time-correlated

single photon counting lifetime spectroscopy system, HAMAMATSU Quantaurus-Tau C11367-02 (for fluorescence lifetime) and C11567-01 (for delayed luminescence lifetime).

TTA-UC and donor phosphorescence quantum yields were measured by using an absolute quantum yield measurement system.³⁸ The sample was held in an integrating sphere and excited by the laser excitation source (532 nm, 200 mW, RGB Photonics). The scattered excitation light was removed using a 532 nm notch filter and emitted light was monitored with a multichannel detector C10027-01 (Hamamatsu Photonics). The spectrometer was calibrated including the integration sphere and notch filter by Hamamatsu Photonics. In general, a quantum yield is defined as the ratio of absorbed photons to emitted photons, and thus the maximum quantum yield (Φ_{UC}) of the bimolecular TTA-UC process is 50%. However, many reports multiply this value by 2 to set the maximum efficiency at 100%. To avoid the confusion between these different definitions, the UC efficiency is written as $\Phi_{UC}' (= 2\Phi_{UC})$ when its maximum is normalized to be 100%.

Sample preparations

To prepare ionic crystals (DCA)₂ADC, 50 μ mol (13.3 mg) of H₂ADC was dissolved in 5.0 mL of methanol, to which 0.1 mmol (20 μ L) of neat dicyclohexylamine was added. Colourless precipitates were immediately formed, and this suspension was left for 3 days at room temperature. The precipitates gradually changed to colourless crystals of (DCA)₂ADC during the incubation. Donor-doped ionic crystals, referred as PdMesoP-(DCA)₂ADC, were prepared by the similar way of (DCA)₂ADC, except for the presence of 0.1 μ mol (67.1 μ g) of PdMesoP in the methanol solution of H₂ADC before adding dicyclohexylamine. Pale-pink crystals of PdMesoP-(DCA)₂ADC were formed during incubation. The size of PdMesoP-(DCA)₂ADC crystals was in the range of a few 100 μ m to 1 mm. As control experiments, crystals with two other compositions were prepared. Nonionic donor PtOEP was used instead of PdMesoP, and the similar crystallization procedure gave PtOEP-(DCA)₂ADC. Nonionic crystals of DPA was doped with PdMesoP by recrystallization of 50 μ mol (16.5 mg) of DPA from hot methanol (30 mL) in the presence of 0.1 μ mol (67.1 μ g) of PdMesoP. All the prepared crystals were collected by suction filtration and washed with methanol, and dried under vacuum at room temperature before measurements. For upconversion measurements, the crystals were placed between two glass plates and sealed in an Ar-filled glove box ($[O_2] < 0.1$ ppm) by using a hot-melt adhesive.

Results and discussion

When a quantitative amount of dicyclohexylamine (0.1 mmol) was added to the 5 ml of 10 mM methanol solution of 9,10-anthracenedicarboxylic acid H₂ADC, colourless precipitates were formed immediately after mixing with dicyclohexylamine. Incubation of this mixture for 3 days at room temperature produced sub-mm sized crystals. FT-IR spectra indicate the formation of the ionic pairs. The C=O vibration band (1676 cm⁻¹) of the carboxylic acid group disappeared, and

replaced by new peaks at 1518 cm⁻¹ and 1398 cm⁻¹ which are assignable to C=O vibration of COO⁻ moiety (Fig. S1).³⁹ Single crystal structural analysis was conducted for this ionic crystal (DCA)₂ADC. The space group was assigned to the P2₁/c monoclinic system. A lamellar structure separated by ionic and nonionic domains was observed (Fig. 2a). In the *b-c* plane, networks of ionic and hydrogen-bond interactions are constructed by carboxylate and ammonium moieties. Within the two-dimensional ionic layers, one-dimensional chains of hydrogen-bonds between ammonium and carboxylate moieties are formed. The edges of anthracene rings are overlapped along the *a*-axis with the nearest C-C distance of 3.6 Å, indicating the existence of a weak π - π interaction (Fig. 2b). In benchmark acceptor 9,10-diphenylanthracene (DPA), phenyl rings provide steric hinderance to avoid the concentration quenching and it shows large Stokes shift associated with the strong π - π interaction. In this work, we employed ADC without phenyl groups as the simplest model, but cyclohexyl moieties effectively serve as spacers to tune the inter-chromophore interactions. The weak inter-chromophore interactions between anthracene moieties are reflected to the optical properties as mentioned below.

UV-vis absorption spectrum of a diluted methanol solution of (DCA)₂ADC (10 μ M) showed π - π^* transition bands with vibronic structures at 337 (0-3), 353 (0-2), 371 (0-1) and 392 nm (0-0) which are characteristic to anthracene-based compounds (Fig. 3). The emission spectrum in methanol shows a vibronic progression mirroring that of the absorption spectra starting with the 0-0 band near 400 nm. In the crystals of (DCA)₂ADC, absorption peaks were moderately broadened and red-shifted

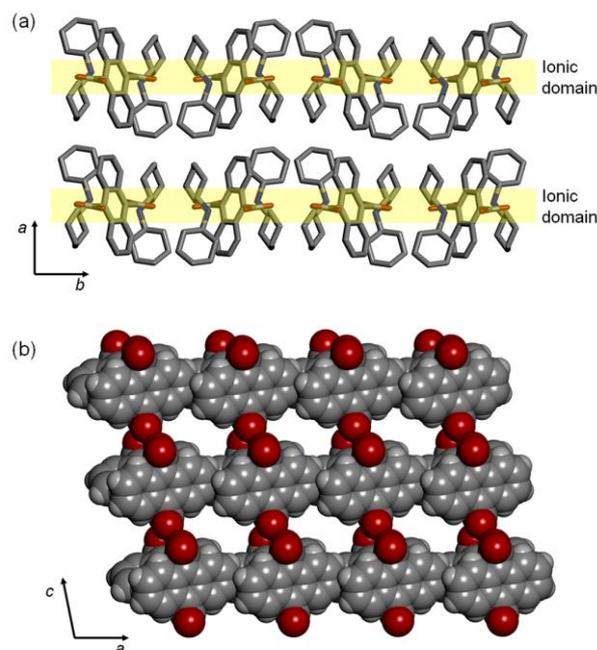


Figure 2. (a) Crystal structure of (DCA)₂ADC viewed along the *c*-axis, showing a lamellar structure consisting of ionic and non-ionic domains. N, blue; O, red; C, grey. Hydrogen atoms are omitted for clarity. (b) Packing structure of ADC. DCA moieties are omitted for clarity. N, blue; O, red; C, grey; H, light grey.

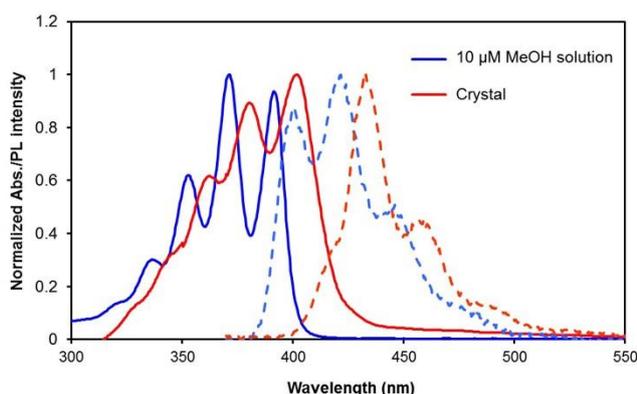


Figure 3. UV-vis absorption spectra (solid lines) and emission spectra (broken lines) of $(DCA)_2ADC$ in a 10 μM methanol solution (blue) and ionic crystals (red). The excitation wavelength was selected as $\lambda_{ex} = 365$ nm.

to 342, 363, 381 and 402 nm. This change reflects weak dipole-dipole interactions between the transition dipole moments of anthracene moieties. A fluorescence spectrum of $(DCA)_2ADC$ crystal showed small red shifts compared to those of the diluted methanol solution of $(DCA)_2ADC$ (Fig. 3). The width of the observed red shift was as small as 75 meV in energy, and such small energy loss is advantageous as emitting materials in sensitized TTA-UC. The reduced 0-0 vibrational band of $(DCA)_2ADC$ crystal emission would be due to the internal filter effect widely observed for condensed solid samples. Interestingly, the ionic crystals $(DCA)_2ADC$ showed a higher fluorescence quantum yield Φ_{FL} of 74% compared to that of $(DCA)_2ADC$ in solution (49%), suggesting the restriction of vibrational deactivation in the rigid crystalline environment. Considering the weak inter-chromophore interactions, a longer fluorescence lifetime of the $(DCA)_2ADC$ crystals (14.5 ns) compared to that in diluted solution (10.2 ns) is probably due to the suppressed non-radiative deactivation in crystals (Fig. S2).

The donor PdMesoP molecules were taken up from the solution to acceptor ionic crystals $(DCA)_2ADC$ during the crystallization process. Pale-pink crystals were obtained after 3 days by incubating the ternary mixture of H_2ADC , dicyclohexylamine and PdMesoP in methanol, suggesting the formation of composite crystals PdMesoP- $(DCA)_2ADC$ (see Experimental section for details). The amount of accommodated donor was estimated by dissolving the composite crystals in methanol and measuring UV-vis absorption spectra. The donor-acceptor molar ratio in PdMesoP- $(DCA)_2ADC$ was estimated as ca. 10000 to 1. Interestingly, the single-crystal X-ray analysis of PdMesoP- $(DCA)_2ADC$ showed that the inclusion of such small amount of donor did not affect the basic acceptor crystal structure (Table S1). To investigate the dispersed state of PdMesoP molecules in the ionic crystal, absorption spectra were measured (Fig. 4). A 10 μM DMF solution of PdMesoP showed a Q(0,0) band at 545 nm, whereas this band is broadened and red-shifted to 558 nm in the bulk PdMesoP solid due to aggregation. Significantly, absorption spectra of PdMesoP- $(DCA)_2ADC$ showed almost similar peaks compared to that in DMF. This result clearly

indicates that PdMesoP molecules are molecularly dispersed without aggregation in the ionic crystals. The inclusion of PdMesoP in $(DCA)_2ADC$ ionic crystals was also evident from the slight decrease of fluorescence quantum yield Φ_{FL} from 74% to 56% which is ascribable to reabsorption and/or energy transfer to in-crystal PdMesoP molecules. This result agrees with a shorter fluorescence lifetime of PdMesoP- $(DCA)_2ADC$ crystals (12.1 ns) compared with that of $(DCA)_2ADC$ crystals (14.5 ns), indicating 17% of acceptor-to-donor singlet back energy transfer (Fig. S3).^{17, 18, 40}

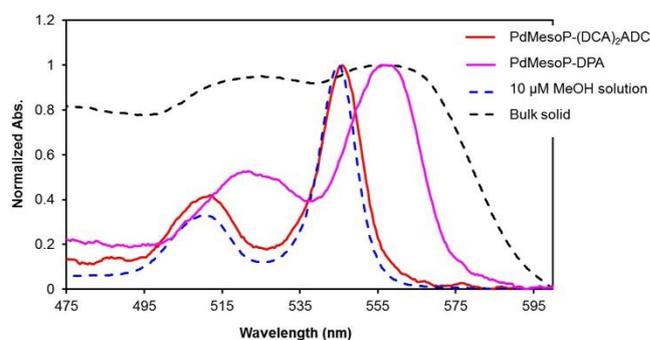


Figure 4. UV-vis absorption spectra of PdMesoP- $(DCA)_2ADC$ crystals (red), PdMesoP-DPA crystals (pink), 10 μM DMF solution of PdMesoP (blue) and bulk PdMesoP (black).

To confirm the role of ionic interactions for molecularly dispersing PdMesoP in $(DCA)_2ADC$ crystals, control experiments were carried out by using nonionic donor PtOEP or nonionic acceptor DPA (see Experimental Section for the sample preparation). When nonionic PtOEP molecules were incorporated into the ionic crystals of $(DCA)_2ADC$, the absorption peaks of PtOEP in the crystals exhibited broadening and red-shift compared to that in molecularly dispersed solution, indicating the aggregation of PtOEP (Fig. S4). Likewise, PdMesoP in DPA crystals showed broadened, red-shifted spectrum and ionic PdMesoP formed aggregates when nonionic acceptor DPA was used as host crystals (Fig. 4). These results indicate the important role of ionic interactions for accommodating donors as monomers in acceptor crystals. That is, the linear ionic networks formed in ionic crystals $(DCA)_2ADC$ show adaptive ability which can alleviate structural mismatch of the incorporated ionic PdMesoP molecules.

The TTA-UC characteristics were evaluated by using sub-mm-sized single crystals of PdMesoP- $(DCA)_2ADC$. The crystals were collected and sealed in an Ar-filled glove box. Under excitation with a 532 nm laser, upconverted emission was clearly observed with the maximum intensity at around 435 nm (Fig. 5). Interestingly, a negligible phosphorescence emission from PdMesoP was observed from PdMesoP- $(DCA)_2ADC$, where the phosphorescence quantum yield (Φ_P) was estimated as less than 0.1%. Taking into account the fact that PdMesoP molecules are molecularly dispersed in the crystals, a 100% TET efficiency from the donor to the surrounding acceptor is strongly suggested. The excitation intensity dependence of UC emission intensity showed a quadratic-to-linear transition by increasing the excitation intensity, characteristic to TTA-based UC

mechanism (Fig. S5).⁴¹⁻⁴³ The crossing point of these two regimes is called as threshold excitation intensity I_{th} , and it represents a useful figure-of-merit of TTA-UC. A relatively low I_{th} value of 49 mW cm^{-2} was observed in PdMesoP-(DCA)₂ADC, reflecting the efficient TET and effective triplet diffusion in ionic crystals.

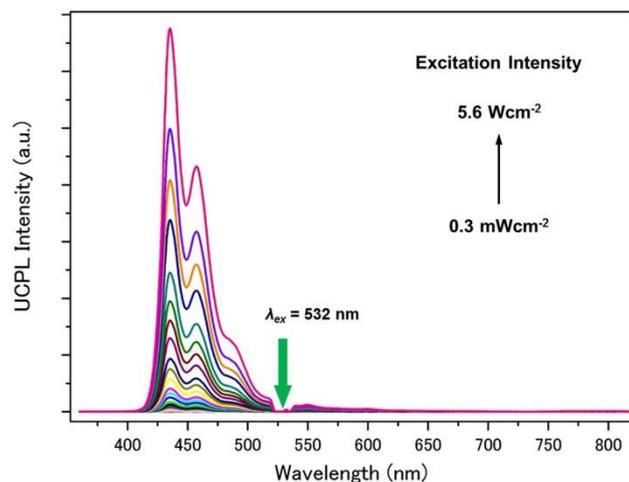


Figure 5. Photoluminescence spectra of PdMesoP-(DCA)₂ADC crystals at various excitation intensities ($\lambda_{ex} = 532$ nm). The scattered incident light was removed by a 532 nm notch filter.

The TTA-UC efficiency Φ_{UC}' of PdMesoP-(DCA)₂ADC was determined by the absolute method using an integrating sphere and the laser excitation source to avoid inaccuracy that could arise from the strong light scattering of the crystals. While the main objective of the current work is to prove the concept using the model ionic crystals, the composite ionic crystals PdMesoP-(DCA)₂ADC already showed a high Φ_{UC}' value of about 6% (Fig. 6). This relatively high Φ_{UC}' value originates from not only the aggregation-free donor accommodation but the less structural defects as discussed below.

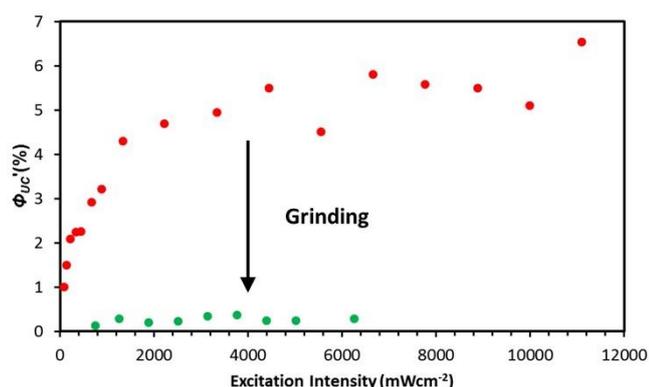


Figure 6. TTA-UC efficiency as a function of the excitation intensity of 532 nm laser for single crystals (red) and ground powders (green) of PdMesoP-(DCA)₂ADC.

To find the clue of the relationship between the UC efficiency and defects, we compared the basic photophysical properties of samples with different degree of the structural

disorder while keeping the identical composition and structure. To introduce defects on purpose, single crystals of PdMesoP-(DCA)₂ADC were mechanically ground using mortar for 10 min in the Ar-filled glove box. PXRD measurements confirmed that this ground powder keeps the crystal structure of (DCA)₂ADC (Fig. S6) Whereas the crystal size became much smaller in the ground powder (sub- μm to μm) compared with the single crystals (a few 100 μm to 1 mm) as observed from scanning electron microscopy (SEM) images (Fig. S7), the absorption and emission spectral features are mostly maintained after the grinding, supporting the intact crystal structure (Fig. S8).

Interestingly, the ground powder of PdMesoP-(DCA)₂ADC showed about 20 times lower UC efficiency Φ_{UC}' (0.3%) than that of original crystals (6%) (Fig. 6). To get insight into this drastic difference in UC efficiency, related parameters were examined. The Φ_{UC}' can be described by the following expression,

$$\Phi_{UC}' = f\Phi_{ISC}\Phi_{ET}\Phi_{TTA}\Phi_{FL} \quad (1)$$

where Φ_{ISC} , Φ_{ET} , and Φ_{TTA} represent the quantum efficiencies of donor ISC, donor-to-acceptor TET, acceptor-acceptor TTA. The parameter f is the statistical probability for obtaining a singlet excited state after the annihilation of two triplet states. In both systems of single crystals and powder, the Φ_{ISC} can be regarded as same. The Φ_{ET} values are ca. 1 in both systems because of no detectable donor phosphorescence peak at 660 nm (Fig. 5 and S9). In general, the parameter f is controlled by a relationship between an energy level of S_1 state and other triplet states.^{42, 44} Considering that the crystal structure and absorption/emission spectra were identical between the single crystals and powder, it is natural to assume that these samples have the similar f value. From these considerations, the differences between the two samples are limited to Φ_{FL} and Φ_{TTA} . We observed a lower Φ_{FL} value for the powder sample (44%) compared with that of single crystals (56%) However, this difference cannot explain the 20 times difference in Φ_{UC}' . These results suggest that a lower Φ_{TTA} value in the powder sample significantly reduces Φ_{UC}' .

In-solution TTA-UC, Φ_{TTA} value can be estimated by fitting UC emission decays with the following equation,^{44, 45}

$$I_{UC}(t) \propto [T_A] = [T_A]_0 \left(\frac{1 - \Phi_{TTA}}{\exp[k_A t] - \Phi_{TTA}} \right)^2 \quad (2)$$

where $I_{UC}(t)$ is the time-dependent UC emission intensity and $[T_A]$ is the population density of acceptor triplets. However, this equation did not fit well the decay curves of the delayed fluorescence from the both of our single crystals and powder, especially at a shorter time range (Fig. S10), implying the presence of deactivation channels not observed in solution. While it is difficult to quantitatively estimate the Φ_{TTA} value, from the tail part of UC emission decays, we observed a shorter triplet lifetime τ_T for the powder sample (1.7 ms) as compared to that for the single-crystal sample (2.9 ms). This comparison gives another support of more prevailing quenching sites in the ground powder. We note that other deactivation pathways not involved in the TTA-UC processes may also exist. The effect of trap sites upon exciton behaviours have been reported for various crystalline systems.^{35-37, 46-48} However, to the best of our knowledge, there have been no reports on the direct evaluation

of defect influence in sensitized TTA-UC. This gives the explanation of the reported poor UC efficiencies in crystalline TTA-UC systems and offers important design guidelines to develop efficient solid-state photon upconverters.

Conclusions

We show that the use of ionic interactions can be the rational strategy to achieve the homogeneous dispersion of ionic donor molecules in acceptor ionic crystals with the maintenance of the highly-ordered structure. In the simple anthracene-based model ionic crystal system, the accommodated anionic donor effectively transfers the triplet energy to the anionic acceptor, resulting in the relatively high UC efficiency. Besides, the effect of defects upon TTA-UC properties was suggested by evaluating TTA-UC properties between two samples, single crystals and mechanically-ground powder. The fundamental knowledge obtained in the current simple model system offers important guidelines for designing upconverting crystals; the formation of ionic networks that adaptively alleviate the structural mismatches of donor molecules and the consequent suppression of disorder. The rational extension of this concept to other conversion wavelength ranges such as near-infrared region with higher fluorescence quantum yield and larger orbital overlaps between neighbouring chromophores would lead to the realization of ultimate solid upconverters exhibiting a close-unity UC efficiency at the solar irradiance.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was partly supported by JSPS KAKENHI Grant Numbers JP25220805, JP17H04799, JP16H06513 (Coordination Asymmetry), JP16H00844 (Soft Molecular Systems), PRESTO program on "Molecular Technology and Creation of New Functions" from JST (JPMJPR14KE), and The Murata Science Foundation.

References

- D. Ginley, M. A. Green and R. Collins, *MRS Bull.*, 2008, **33**, 355-364.
- W. Shockley and H. J. Queisser, *J. Appl. Phys.*, 1961, **32**, 510-519.
- W. Kaiser and C. G. B. Garrett, *Phys. Rev. Lett.*, 1961, **7**, 229-8.
- M. Pawlicki, H. A. Collins, R. G. Denning and H. L. Anderson, *Angew. Chem. Int. Ed.*, 2009, **48**, 3244-3266.
- F. Auzel, *Chem. Rev.*, 2004, **104**, 139-173.
- W. Q. Zou, C. Visser, J. A. Maduro, M. S. Pshenichnikov and J. C. Hummelen, *Nat. Photon.*, 2012, **6**, 560-564.
- S. Balushev, T. Miteva, V. Yakutkin, G. Nelles, A. Yasuda and G. Wegner, *Phys. Rev. Lett.*, 2006, **97**, 143903.
- T. N. Singh-Rachford and F. N. Castellano, *Coordin. Chem. Rev.*, 2010, **254**, 2560-2573.
- J. Z. Zhao, S. M. Ji and H. M. Guo, *Rsc Adv.*, 2011, **1**, 937-950.
- A. Monguzzi, R. Tubino, S. Hoseinkhani, M. Campione and F. Meinardi, *Phys. Chem. Chem. Phys.*, 2012, **14**, 4322-4332.
- Y. C. Simon and C. Weder, *J. Mater. Chem.*, 2012, **22**, 20817-20830.
- J. H. Kim and J. H. Kim, *J. Am. Chem. Soc.*, 2012, **134**, 17478-17481.
- V. Gray, D. Dzebo, M. Abrahamsson, B. Albinsson and K. Moth-Poulsen, *Phys. Chem. Chem. Phys.*, 2014, **16**, 10345-10352.
- J. Zhou, Q. Liu, W. Feng, Y. Sun and F. Y. Li, *Chem. Rev.*, 2015, **115**, 395-465.
- M. Haring, R. Perez-Ruiz, A. Jacobi von Wangelin and D. D. Diaz, *Chem. Commun.*, 2015, **51**, 16848-16851.
- T. F. Schulze and T. W. Schmidt, *Energy Environ. Sci.*, 2015, **8**, 103-125.
- N. Yanai and N. Kimizuka, *Acc. Chem. Res.*, 2017, **50**, 2487-2495.
- S. P. Hill and K. Hanson, *J. Am. Chem. Soc.*, 2017, **139**, 10988-10991.
- Z. Y. Huang and M. L. Tang, *J. Am. Chem. Soc.*, 2017, **139**, 9412-9418.
- R. Vadrucci, A. Monguzzi, F. Saenz, B. D. Wilts, Y. C. Simon and C. Weder, *Adv. Mater.*, 2017, **29**.
- R. R. Islangulov, J. Lott, C. Weder and F. N. Castellano, *J. Am. Chem. Soc.*, 2007, **129**, 12652-12653.
- J. H. Kim, F. Deng, F. N. Castellano and J. H. Kim, *Chem. Mater.*, 2012, **24**, 2250-2252.
- F. Marsico, A. Turshatov, R. Pekoz, Y. Avlasevich, M. Wagner, K. Weber, D. Donadio, K. Landfester, S. Balushev and F. R. Wurm, *J. Am. Chem. Soc.*, 2014, **136**, 11057-11064.
- S. H. Lee, D. C. Thevenaz, C. Weder and Y. C. Simon, *J. Polym. Sci. A Polym. Chem.*, 2015, **53**, 1629-1639.
- A. Monguzzi, M. Mauri, A. Bianchi, M. K. Dibbanti, R. Simonutti and F. Meinardi, *J. Phys. Chem. C*, 2016, **120**, 2609-2614.
- A. Monguzzi, A. Oertel, D. Braga, A. Riedinger, D. K. Kim, P. N. Knusel, A. Bianchi, M. Mauri, R. Simonutti, D. J. Norris and F. Meinardi, *ACS applied materials & interfaces*, 2017, **9**, 40180-40186.
- S. Balushev, V. Yakutkin, G. Wegner, B. Minch, T. Miteva, G. Nelles and A. Yasuda, *J. Appl. Phys.*, 2007, **101**, 023101.
- R. Vadrucci, C. Weder and Y. C. Simon, *J. Mater. Chem. C*, 2014, **2**, 2837-2841.
- H. Goudarzi and P. E. Keivanidis, *J. Phys. Chem. C*, 2014, **118**, 14256-14265.
- R. Andemach, H. Utzat, S. D. Dimitrov, I. McCulloch, M. Heeney, J. R. Durrant and H. Bronstein, *J. Am. Chem. Soc.*, 2015, **137**, 10383-10390.
- M. Hosoyamada, N. Yanai, T. Ogawa and N. Kimizuka, *Chem. - Eur. J.*, 2016, **22**, 2060-2067.
- N. Yanai and N. Kimizuka, *Chem. Commun.*, 2017, **53**, 655-655.
- K. Kamada, Y. Sakagami, T. Mizokuro, Y. Fujiwara, K. Kobayashi, K. Narushima, S. Hirata and M. Vacha, *Mater. Horiz.*, 2017, **4**, 83-87.
- T. Ogawa, N. Yanai, H. Kouno and N. Kimizuka, *J. Photon. Energy*, 2017, **8**, 022003.
- O. V. Mikhnenko, J. Lin, Y. Shu, J. E. Anthony, P. W. M. Blom, T. Q. Nguyen and M. A. Loi, *Phys. Chem. Chem. Phys.*, 2012, **14**, 14196-14201.
- O. V. Mikhnenko, M. Kuik, J. Lin, N. van der Kaap, T. Q. Nguyen and P. W. M. Blom, *Adv. Mater.*, 2014, **26**, 1912-1917.
- J. D. A. Lin, O. V. Mikhnenko, T. S. van der Poll, G. C. Bazan and T. Q. Nguyen, *Adv. Mater.*, 2015, **27**, 2528-2532.
- P. F. Duan, N. Yanai, H. Nagatomi and N. Kimizuka, *J. Am. Chem. Soc.*, 2015, **137**, 1887-1894.

39. A. Dawn, N. Fujita, S. Haraguchi, K. Sada and S. Shinkai, *Chem. Commun.*, 2009, **0**, 2100-2102.
40. J. C. Wang, S. P. Hill, T. Dilbeck, O. O. Ogunsolu, T. Banerjee and K. Hanson, *Chem. Soc. Rev.*, 2018, **47**, 104-148.
41. A. Monguzzi, J. Mezyk, F. Scotognella, R. Tubino and F. Meinardi, *Phys. Rev. B*, 2008, **78**, 195112.
42. Y. Y. Cheng, T. Khoury, R. G. C. R. Clady, M. J. Y. Tayebjee, N. J. Ekins-Daukes, M. J. Crossley and T. W. Schmidt, *Phys. Chem. Chem. Phys.*, 2010, **12**, 66-71.
43. A. Haefele, J. Blumhoff, R. S. Khayzer and F. N. Castellano, *J. Phys. Chem. Lett.*, 2012, **3**, 299-303.
44. Y. Y. Cheng, B. Fuckel, T. Khoury, R. G. C. R. Clady, M. J. Y. Tayebjee, N. J. Ekins-Daukes, M. J. Crossley and T. W. Schmidt, *J. Phys. Chem. Lett.*, 2010, **1**, 1795-1799.
45. A. Monguzzi, F. Bianchi, A. Bianchi, M. Mauri, R. Simonutti, R. Ruffo, R. Tubino and F. Meinardi, *Adv. Energy Mater.*, 2013, **3**, 680-686.
46. K. Yokoi and Y. Ohba, *Chem. Phys. Lett.*, 1986, **129**, 240-243.
47. B. Manna, R. Ghosh and D. K. Palit, *J. Phys. Chem. C*, 2015, **119**, 10641-10652.
48. C. Grieco, G. S. Doucette, R. D. Pensack, M. M. Payne, A. Rimshaw, G. D. Scholes, J. E. Anthony and J. B. Asbury, *J. Am. Chem. Soc.*, 2016, **138**, 16069-16080.