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Enhanced xenon adsorption and separation with an anionic indium–organic framework by ion exchange with Co^{2+} †

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The separation of xenon/krypton is industrially significant and an environmental concern. Adsorptive capture and separation xenon from krypton using porous MOFs provides an energy and capital efficient approach compared with the current cryogenic distillation process. Herein, we investigated the adsorptive Xe/Kr separation potential of three anion In-MOFs (CPM-5, CPM-6 and the Co^{2+} -exchanged framework analogue Co^{2+} -CPM-6). Anionic In-MOF Co^{2+} -CPM-6 with Co^{2+} ions in pore spaces has been obtained using a simple cation-exchange process and exhibits much higher Xe adsorption capacity and Xe/Kr selectivity than organic cation-analogues CPM-5 ($[(\text{CH}_3)_2\text{NH}_2]^+$) and CPM-6 ($[(\text{CH}_3\text{NH}_3)]^+$), verified by single-component gas isotherms, IAST calculations and breakthrough experiments. The enhanced adsorptive Xe/Kr separation performance for Co^{2+} -CPM-6 could be due to the increased pore size or accessible micropore volume and enhanced electric field within the pore spaces, which could induce strong interaction with Xe and simultaneously reduce the affinity with Kr.

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Introduction

Rare or noble gases xenon (Xe) and krypton (Kr) are commonly encountered in low concentrations (Xe exists in the atmosphere at 0.087 ppmv, and krypton (Kr) at 1.14 ppmv) and their extraction is consequently difficult.¹ However, these noble gases are industrially valuable and have many significant uses including commercial lighting, medical imaging, insulation and lasers.^{1,2} On the other hand, radioisotopes of xenon and krypton (^{85}Kr and ^{133}Xe) are released as by-products from nuclear reprocessing plants. For environmental concerns and recycling of noble gases, the long-lived and hazardous ^{85}Kr (half life of 10.8 years) should be separated and captured from effluent gas mixtures of nuclear fuel reprocessing plants.^{3,4} Meanwhile, separation and capture of xenon from radioactive krypton in the off-gas stream after a short-time decay of the short half-life xenon isotope would provide a new resource of xenon for industrial use. Currently, a 20/80 molar mixture of Xe–Kr obtained as a by-product in cryogenic air separation must undergo further cryogenic distillation to produce pure xenon and krypton. And cryogenic process is also suggested for the removal of radioactive noble gases from off-gas streams in future nuclear reprocessing plants. However, this process is

highly energy and capital intensive.⁵ Therefore, effective capture and separation of Xe–Kr at room temperature becomes an urgent and significant issue to be resolved.

Adsorptive separation of Xe–Kr using porous materials at room temperature could be a suitable alternative to cryogenic process for its easy operation and low costs. Thus, adsorbents with high Xe/Kr selectivity and capacity are essential for the design of adsorptive separation process such as pressure swing adsorption (PSA) and vacuum swing adsorption (VSA).⁶ In this regard, zeolites and active carbon have been used and evaluated for Xe/Kr separation at ambient conditions.^{7–9} However, their low Xe capacity and selectivity limits their practical use. Metal–organic frameworks (MOFs), a new kind of hybrid crystalline porous material which are hybrid lattices of organic ligands and metal nodes (metal ions or clusters), have attracted enormous attention over the past two decades for its ultrahigh specific surface area, tunable pore structure, adjustable chemical environment and high thermal stability.^{10–12} Owing to this feature, MOFs appear to be promising as cost-effective and efficient adsorbents for gas storage and separation.^{13–21} Recently, the effective separation of noble gases Xe and Kr using MOFs has received much more attention and become a new area of focus.^{22–36}

Up to date, studies on Xe/Kr separation using MOFs are still scarce.³⁷ Wang *et al.* showed that $\text{Co}_3(\text{HCOO})_6$ can commensurately adsorb Xe and high Xe/Kr selectivity due to the unique pore size and channel shape.³⁸ Thallapally *et al.* reported that NiDOBDC adsorbs more Xe compared to activated carbon and also showed that silver loading within NiDOBDC notably

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enhances both Xe capacity and Xe/Kr selectivity.^{39,40} Cooper *et al.* reported that a porous organic cage CC3, though not a MOF, shows significantly high Xe capacity and Xe/Kr selectivity at low concentrations.⁴¹ Xiong *et al.* reported a flexible zinc tetrazolate framework [Zn(mtzt)₂] which exhibits breathing behaviour on Xe adsorption and has a high Xe/Kr selectivity measured by breakthrough experiment using binary Xe–Kr gas mixture.⁴² Recently, Thallapally *et al.* reported that SBMOF-1 exhibits by far the highest reported xenon adsorption capacity and a remarkable Xe/Kr selectivity under conditions pertinent to nuclear fuel reprocessing.⁴³ Computational screening studies carried by Snurr *et al.* suggest that MOFs with cylindrical pores just large enough to fit a single xenon atom would have the highest Xe/Kr selectivity but better materials remains to be discovered.⁴⁴

Recent studies have indicated several features that are desirable for enhancing adsorption capacity and selectivity of xenon at room temperature. One feature is the suitable pore size commensurate with the size of Xe molecule (4.1 Å). Another factor currently receiving considerable attention is introducing individual localized binding sites using methods such as (1) deposition of Ag clusters in the porous MOFs and (2) creation of unsaturated metal sites (also called open-metal sites) and more polarizing organic groups (–I, –OH).^{40,45–47} Furthermore, the porous solid-Xe molecule interactions can also be enhanced by employing the electric field across the pore space that is capable of polarizing Xe molecules and increasing the interaction with charged internal pore surface or charged extraframework species.⁴⁰ However, this approach is rarely studied because most of MOFs exhibit an electrically neutral framework. Thus, it is essential and meaningful to investigate the Xe adsorption and separation properties of anionic MOFs and the influence of cations on their separation performance. Indium(III) often exhibits an eight-coordinated central building block with four equivalent carboxylate ligands and has been frequently used to design and synthesis of anionic MOFs with low connectivity and high surface areas. Two anionic In-MOF: [(CH₃)₂NH₂][In₃O(BTC)₂(H₂O)₃]₂[In₃(BTC)₄·solvent (denoted as CPM-5, CPM = crystalline porous materials) and [CH₃NH₃][In₃O(BTC)₂(H₂O)₃]₂[In₃(BTC)₄·solvent (CPM-6, BTC = 1,3,5-benzenetricarboxylate) firstly reported by Zheng *et al.* exhibit high surface areas, good hydrothermal stability, coupled with suitable pore size (4–6 Å) and strong charge-induced forces within the internal pore space which render them good candidates for the study of Xe selective sorption properties.

In this work, we firstly describe facile separation of Xe from Kr at room temperature using three anionic microporous MOFs based upon In(III) ions, CPM-5, CPM-6 and the Co²⁺-exchanged framework analogue Co²⁺-CPM-6. More smaller and higher polar Co²⁺ ions could be easily introduced into the pores of the anionic microporous framework using a simple ion-exchange with organic cations ([CH₃NH₃]⁺), which provides an approach to tune the pore volume and electric field across the pore space. Experimental adsorption isotherms and breakthrough curves under packed-bed mixture flow conditions indicates that Co²⁺-CPM-6 exhibits much higher Xe adsorption capacity and Xe/Kr

selectivity compared with its organic cation-analogues CPM-5 ([CH₃)₂NH₂]⁺) and CPM-6 ([CH₃NH₃]⁺).

Experimental

Materials and measurements

All reagents and solvents were used as received from commercial suppliers without further purification. Thermogravimetric analyses (TGA) were performed on a TA TGA/Q500 analyzer with heating rate of 5 °C min^{−1} from 27 to 800 °C. Powder X-ray diffraction (PXRD) patterns were recorded by a Bruker D8-advance Powder X-ray Diffractometer Instrument operated at 40 kV and 44 mA with a scan of 1.0 deg min^{−1}. Analyses for the concentrations of Co²⁺, In³⁺ were carried out on an Inductively Coupled Plasma-Optical Emission Spectroscopy (ICP-OES) analyzer. Calibration curves for ICP-OES were prepared by dilution of commercially available standards with the sample dissolves in concentrated HCl, and diluted to proper concentration for measurement. The verification of Co²⁺ in CPM-6 was also carried out on a Perkin-Elmer PHI-5702 multi-functional X-ray photoelectron spectroscopy (XPS) with Al K α radiation.

Preparation of CPM-5 and CPM-6

CPM-5 was synthesized as a slight modification of the procedure of ref. 49, in which a mixture of 0.2 g In(NO₃)₃·5H₂O and 0.17 g H₃BTC was stirred in a mixed solution of H₂O/DMF (1.0 g/4.0 g) for 1 hour, and then transferred to the Teflon-lined stainless steel autoclave (20 ml), and kept at 120 °C for 5 days. CPM-6 was synthesized using the same procedure in which the DMF was replaced by NMF. After washed by methanol several times, the colorless crystals were obtained.

Preparation of Co²⁺-CPM-6

The as-prepared CPM-6 sample was ion-exchanged with Co²⁺, immersed in methanol containing 0.1 M Co(NO₃)₂ and changed with fresh new Co(NO₃)₂ methanol solution every day for four days. And then the samples were filtered and washed with methanol several times to obtain Co²⁺ ions exchanged CPM-6 designated as Co²⁺-CPM-6.

Gas adsorption measurements

Xe, Kr, Ar, N₂ adsorption and desorption isotherms were measured on the Micromeritics 3Flex Surface Characterization Analyzer. To get the solvent free framework, the fresh samples of CPM-5 and CPM-6 were guest-exchanged with dry methanol 4 times per day for 2 days, filtered and degassed at ambient temperature for 6 h, and then vacuumed at 180 °C overnight to measurements. The fresh samples of Co²⁺-CPM-6 were filtered and washed by methanol several times and degassed at ambient temperature for 6 h, and then vacuumed at 180 °C overnight to measurements. The adsorption isotherms at 77 K were measured upon the liquid nitrogen bath. The temperatures of adsorption isotherms at 273 K and 298 K were maintained using a Micromeritics's ISO Controller (Sub-Ambient, Thermoelectric Cooled Dewar). Ultrahigh-purity-grade (>99.999%) Xe, Kr, Ar, N₂ and He gases were applied in the adsorption measurements.



Results and discussion

Characterization of CPM-5, CPM-6 and Co²⁺-CPM-6

CPM-5 and CPM-6 were synthesized using a slight modification of a published procedure.⁴⁹ The phase purity of the sample was confirmed by PXRD. CPM-5 and CPM-6 firstly reported by Zheng *et al.* have the same framework structure just with different organic cations ($[(CH_3)_2NH_2]^+$ for CPM-5, $CH_3NH_3^+$ for CPM-6). As shown in Fig. 1, CPM-5 and CPM-6 have a fascinating cage-within-cage architecture structure. The bigger In_{24} sodalite cage built with negatively charged single-metal building blocks $\{In(CO_2)_4\}^-$ and BTC^{3-} ligands with $25.7 \times 25.7 \times 25.7 \text{ \AA}^3$ in dimension is far too large for capture of Xe atoms. The small inner In_{12} cage comprised of four positively charged trimeric clusters $\{In_3O\}^+$ and three BTC^{3-} ligands with a dimension of 8.7 Å. The In_{12} cage was interconnected with the In_{24} cage by 12 BTCs ligands which segment the pore space of the large sodalite cage into multiple domains. Drifting cations – $(CH_3)_2NH_2^+$ (CPM-5) and $CH_3NH_3^+$ (CPM-6) are amid these networks. The whole pore space partitioned by BTC linkers and occupied by cations exhibits microporous pore network structure. The micropore size of 4–6 Å was calculated using Horvath–Kawazoe method. Additionally, removal of the three dangling water on $\{In_3O\}(O_2CR)_6(H_2O)_3\}$ of the inner In_{12} cage during the activation procedure generates In^{3+} sites being exposed to pore surface for Xe adsorption.⁴⁸ Interestingly, the outer In_{24} cage composed of $\{In(CO_2)_4\}^-$ units is negatively charged while the inner In_{12} cage made from $\{In_3O\}(O_2CR)_6(H_2O)_3\}$ units is positively charged, resulting in an electric field within the pore space.⁴⁹ What's more, the organic cations within the pores could be easily exchanged by metal ions to tune the pore size and electric field for Xe adsorption.⁵⁰ In this work, we choose Co^{2+} ions to exchange with $[CH_3NH_3]^+$ using the as-synthesized CPM-6 samples based on several reasons: (1) bivalent Co^{2+} ions having higher polarizing ability compared to $[CH_3NH_3]^+$ and bivalent metal ion would enhance electric field within the pore

space, (2) for M-MOF-74 ($M = Co^{2+}, Mg^{2+}, Zn^{2+}$ and Ni^{2+}), Co^{2+} shows significantly stronger binding to Xe molecules than Mg^{2+}, Zn^{2+} and Ni^{2+} , which has been proved by experimental and simulated results,^{45,51,52} (3) Co^{2+} is smaller than $[CH_3NH_3]^+$, consequently releasing more space volume for Xe adsorption.

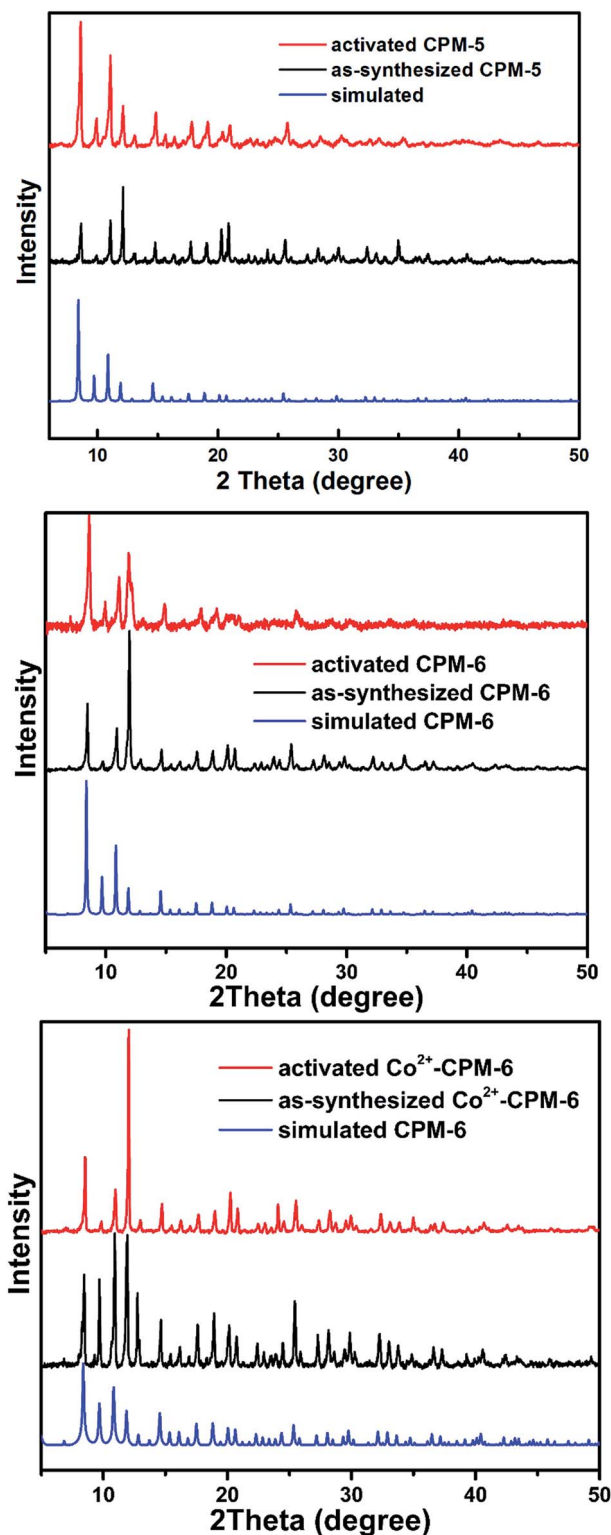


Fig. 2 The PXRD patterns of CPM-5, CPM-6, and Co²⁺-CPM-6.

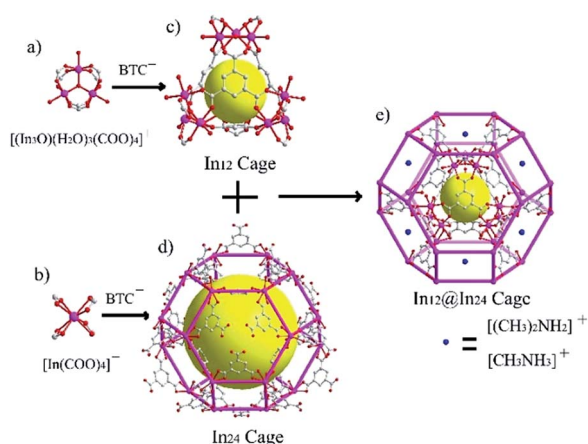


Fig. 1 (a–e) Structures of trimeric $\{(In_3O)(H_2O)_3\}$ unit, monomeric In^{3+} ion, In_{12} cage, In_{24} cage and $In_{12}@In_{24}$ cage, respectively. The purple solid lines represent BTCs. The blue solid balls represent the cations ($[(CH_3)_2NH_2]^+$, $[CH_3NH_3]^+$). The small yellow ball represents the In_{12} cage cavity. The big yellow ball represents In_{24} cage cavity.



Co²⁺-exchange sample, Co²⁺-CPM-6 was obtained by soaking as-synthesized CPM-6 samples into a methanol solution of Co(NO₃)₂. In³⁺/Co²⁺ molar ratio in Co²⁺-CPM-6 measured by ICP-OES is 29, which means 62% [CH₃NH₃]⁺ cations has been replaced by Co²⁺ ions. And XPS result of Co²⁺-CPM-6 (see Fig. S13†) also showed that Co²⁺ ions were successfully introduced into the framework.

Powder X-ray diffraction patterns indicate that pure phase samples of CPM-5 and CPM-6 have been successfully synthesized (shown in Fig. 2). Co²⁺-CPM-6 nearly has the same PXRD patterns with CPM-6, which shows that Co²⁺-CPM-6 maintains its framework structure and crystalline state well during the Co²⁺-exchange process. And the PXRD patterns for the activated samples illustrate that CPM-5, CPM-6 and Co²⁺-CPM-6 are stable enough for gas sorption measurements after their activation procedure.

As shown in Fig. 3, N₂ sorption isotherms of CPM-5, CPM-6 and Co²⁺-CPM-6 at 77 K indicate that all three samples exhibit a typical reversible type-I behavior. We adopted two models: Langmuir and BET surface area to precisely evaluate the specific surface areas of these three materials. The Langmuir and BET surface areas of CPM-5 are 768 m² g⁻¹ and 521 m² g⁻¹, respectively. And the micropore volume of CPM-5 (using *t*-plot method) was also calculated to be 0.242 cm³ g⁻¹. The (Langmuir) BET surface areas and micropore volume of CPM-6 were calculated to be (871 m² g⁻¹) 583 m² g⁻¹ and 0.282 cm³ g⁻¹, which are higher than that of CPM-5. This could be due to small cation size of [CH₃NH₃]⁺ in CPM-6 taking up less pore space compared with [(CH₃)₂NH₂]⁺ in CPM-5. The surface area and micropore volume of CPM-5 and CPM-6 in this work are consistent with the reported literature.⁴⁹ Co²⁺-CPM-6 with the small Co²⁺ ions for charge-compensating exhibits the highest (Langmuir) BET surface areas and micropore volume compared with CPM-5 and CPM-6 and were calculated to be (1055) 671 m² g⁻¹ and 0.305 cm³ g⁻¹, respectively. Langmuir model has a relative higher value than BET model. The differences between Langmuir and BET surface areas could be attributed to

the different hypothesis that Langmuir model assumes monolayer adsorption while BET model assumes multilayer adsorption. Thus it is a reasonable result that Langmuir model has a higher value than BET model. However, the specific surface areas of these two models don't differ too much, which manifests that CPM-5, CPM-6 and Co²⁺-CPM-6 belong to the micropore materials and the pore sizes won't exceed 7 Å (see Fig. S3†). The DFT pore sizes distribution calculated by DFT model shows

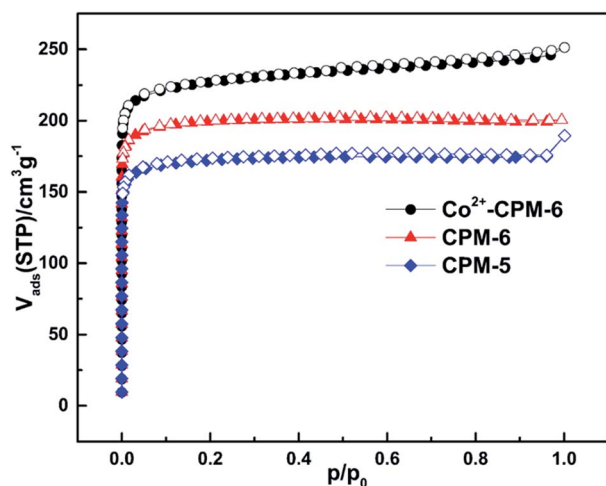


Fig. 3 N₂ sorption isotherms of CPM-5, CPM-6 and Co²⁺-CPM-6 at 77 K.

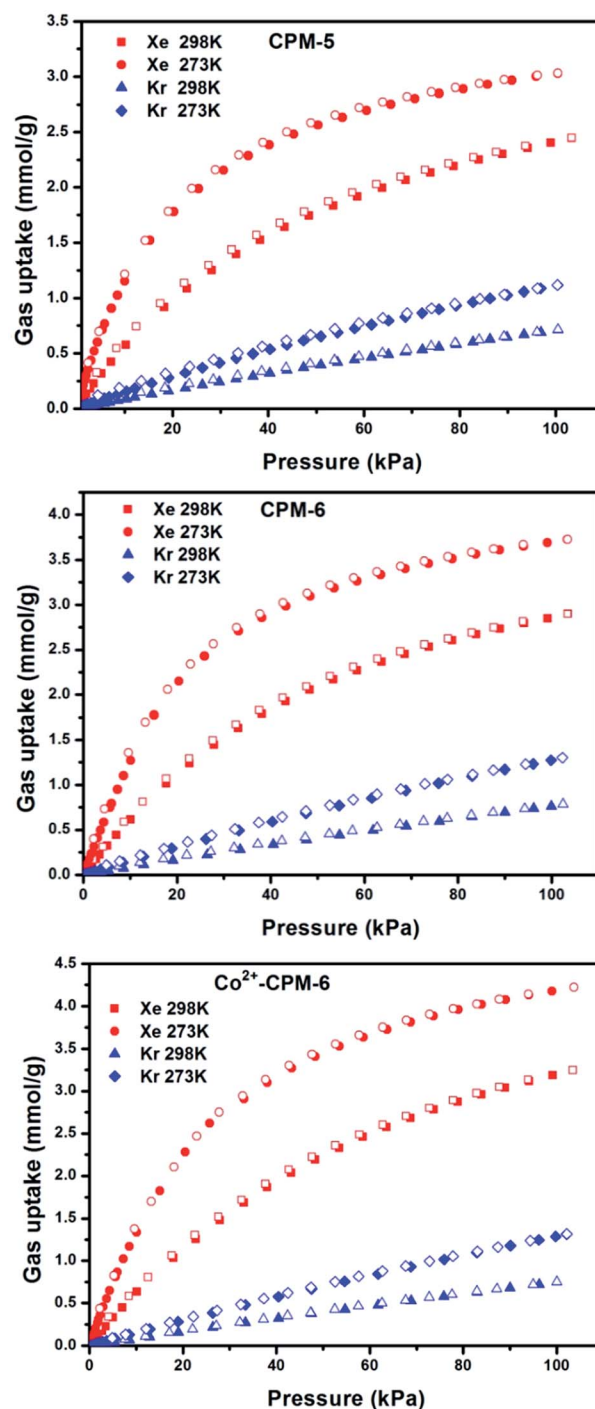


Fig. 4 Xe and Kr adsorption and desorption isotherms of CPM-5, CPM-6 and Co²⁺-CPM-6 at 273 K and 298 K.



Table 1 Gas uptake (mmol g^{-1}) at 298 K and 1 bar, Henry constants ($\text{mmol g}^{-1} \text{bar}^{-1}$) and Q_{st} (kJ mol^{-1}) on CPM-5, CPM-6 and Co^{2+} -CPM-6^a

	N_{Xe} (mmol g^{-1})	N_{Kr} (mmol g^{-1})	$N_{\text{Xe}}/N_{\text{Kr}}$	H_{Xe} ($\text{mmol g}^{-1} \text{bar}^{-1}$)	H_{Kr} ($\text{mmol g}^{-1} \text{bar}^{-1}$)	$H_{\text{Xe}}/H_{\text{Kr}}$	$Q_{\text{st}}(\text{Xe})$	$Q_{\text{st}}(\text{Kr})$
CPM-5	2.44	0.71	3.1	6.06	0.89	6.8	25.0	17.3
CPM-6	2.89	0.78	3.7	6.15	0.85	7.2	25.1	17.1
Co^{2+} -CPM-6	3.20	0.75	4.3	6.28	0.81	7.8	25.9	16.6

^a N_{Xe} and N_{Kr} represent Xe and Kr uptake, respectively; H_{Xe} and H_{Kr} represent Henry's constant of Xe and Kr.

that Co^{2+} -CPM-6 exhibits a little decrease of pore space in the range of 5.4–6.4 Å and a distinct increase of pore space in the range of 6.4–8.6 Å compared with CPM-6. These results show that ion-exchange with small metal ions in anionic framework could provide an effective way to enhance the specific surface area and adjust the micropore volume or pore size (see Fig. S3†).

Single component sorption isotherms, Henry's constants and isosteric heats of adsorption

The establishment of the permanent porosity enabled us to examine their gas sorptions of Xe and Kr. As shown in Fig. 4. Pure adsorption and desorption isotherms for Xe and Kr in CPM-5, CPM-6 and Co^{2+} -CPM-6 were measured at 298 K and 273 K up to 1 bar. Adsorption and desorption isotherms for Xe and Kr are fully reversible and the sorption of Xe exhibits type-I behavior, which was typical for strong adsorbates sorption in microporous materials. The uptake of Xe at 298 K and 1 bar for CPM-5 and CPM-6 are 2.44 mmol g^{-1} and 2.89 mmol g^{-1} , 3.4 times and 3.7 times higher than their Kr uptakes (0.71 mmol g^{-1} for CPM-5 and 0.78 mmol g^{-1} for CPM-6). The Xe uptake of Co^{2+} -CPM-6 at 298 K and 1 bar is 3.20 mmol g^{-1} , which is 4.3 times higher than its Kr uptake (0.75 mmol g^{-1}). It should be noted that Co^{2+} -CPM-6 has an obvious higher Xe uptake than CPM-6 but a slightly low Kr uptake compared with CPM-6. The enhancement of Xe adsorption capacity for Co^{2+} -CPM-6 could be due to an increase in the accessible micropore volume. The slightly lower Kr uptake of Co^{2+} -CPM-6 may be due to the decrease of micropore volumes in the range of 5.4–6.4 Å, that are suitable for Kr adsorption.

In order to understand and explain the different performances for these three MOFs on uptake of Xe over Kr, we calculated the Henry's constants based on single component sorption isotherms to evaluate the gas-adsorbent interactions. Henry's constant is a useful measure of adsorbents' affinity for adsorbates, since it represents the partition of the adsorbate between its bulk phase and adsorbed phase at very low pressures.³⁸ The Henry's constants for Xe and Kr in CPM-5, CPM-6 and Co^{2+} -CPM-6 were calculated using low-pressure isotherms at 298 K. We fitted the low-pressure isotherms which passed through the original point and appeared linear fit, and obtained slopes of these fitting lines which represent the Henry's constants (see Fig. S6–S11†). As shown in Table 1, the order of the calculated Henry's constants for Xe is Co^{2+} -CPM-6 > CPM-6 > CPM-5. Interestingly, the order of the calculated Henry's constant for Kr is CPM-5 > CPM-6 > Co^{2+} -CPM-6, which is opposite to the order of the Henry's constants for Xe. The enhancement of Henry's constant for Xe in Co^{2+} -CPM-6 could

be due to the more polarized electric field across the pore space or strong interaction between Co^{2+} ions and Xe molecules. The lower Henry's constant for Kr in Co^{2+} -CPM-6 is main due to the decreased pore volumes in the range of 5.4–6.4 Å that are just fit for adsorption of Kr atoms. To verify adsorption strength of Xe and Kr in CPM-5, CPM-6 and Co^{2+} -CPM-6, we calculated the isosteric heats of adsorption (Q_{st}) from single component adsorption isotherms as a function of quantity adsorbed at different temperatures (273 K and 298 K) using the Clausius–Clapeyron equation. As shown in Fig. 5, Xe Q_{st} values of CPM-5 and CPM-6 at low loadings are 25.0 kJ mol^{-1} and 25.1 kJ mol^{-1} , which are nearly the same. Co^{2+} -CPM-6 has a higher Xe Q_{st} value (25.9 kJ mol^{-1}) than CPM-5 and CPM-6. The Kr Q_{st} values of these three MOFs at low loading are as follows: CPM-5 (17.3 kJ mol^{-1}) > CPM-6 (17.1 kJ mol^{-1}) > Co^{2+} -CPM-6 (16.6 kJ mol^{-1}), which matches well with the order of the Henry's constants for Kr. All these results indicates that introduce of small and polar Co^{2+} ions could enhance the Xe interaction with pore surface and reduce affinity of Kr. The Xe/Kr selectivity calculated by the ratio of Henry's constants follows the order: Co^{2+} -CPM-6 (7.8) > CPM-6 (7.2) > CPM-5 (6.8), which indicates that Co^{2+} -CPM-6 has the strongest interaction with Xe molecules of these three MOFs.

IAST selectivity

Ideal adsorbed solution theory (IAST) is a thermodynamic method which can predict the selectivity of gas mixtures based

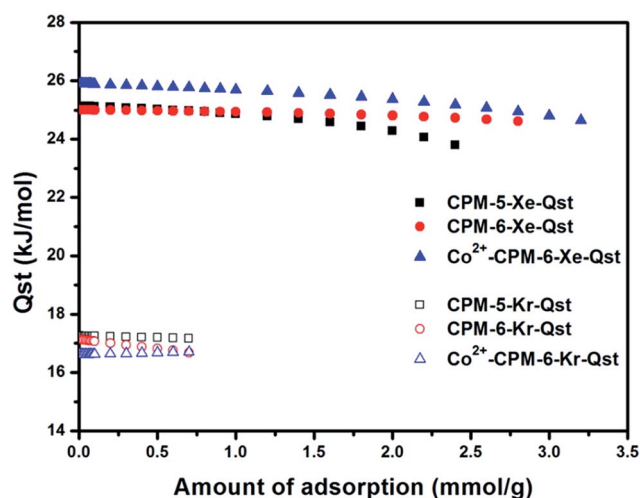


Fig. 5 The isosteric heats of adsorption (Q_{st}) for Xe and Kr in CPM-5, CPM-6 and Co^{2+} -CPM-6.



on the experimental single component adsorption isotherms and has been proved to be an effective approach to evaluate the separation ability of various materials.^{53–55} In order to obtain good prediction by IAST, pure isotherms of Xe and Kr were fitted using the dual-site Langmuir–Freundlich (DSLFL) model (see Table S1†).⁵⁶ In this work, we predicted the Xe/Kr selectivity by IAST for 20/80 Xe/Kr gas mixture, which is an industrially relevant composition for Xe–Kr gas mixture to assess the potential of CPM-5, CPM-6 and Co²⁺-CPM-6 on Xe/Kr separation. As shown in Fig. 6, Co²⁺-CPM-6 exhibits a distinct enhancement on Xe/Kr selectivity compared with CPM-6 after the Co²⁺-exchange process. The IAST Xe/Kr selectivities of these three materials follow the order of Co²⁺-CPM-6 (9.3) > CPM-6 (7.3) > CPM-5 (7.1) at 100 kPa and 298 K, which agrees with the results from Henry's constants. This could be attributed to the stronger electric field across the pore space with Co²⁺ ions as compensating cations in Co²⁺-CPM-6, that is capable of polarizing Xe atoms and increasing its interaction with charged pore surface or Co²⁺ ions. The IAST Xe/Kr selectivity of Co²⁺-CPM-6 is among the reported high Xe/Kr selectivity porous materials: SBMOF-1 (16), SBMOF-2 (10), Co₃(HCOO)₆ (11), CC3 (12.5), Ag@MOF-74Ni (11.5), Co-MOF-74 (12).^{43,46}

Dynamic breakthrough experiments

We also carried out the breakthrough experiments at room temperature to evaluate the potential of these three MOFs for the adsorptive separation of Xe/Kr mixtures, under the real world conditions (see Fig. S12†). Fig. 7 shows the breakthrough curves of Xe and Kr upon separation of a 20/80 Xe/Kr mixture on the column packed with CPM-5, CPM-6 and Co²⁺-CPM-6 at 298 K and 1 bar. For all three MOFs, Kr firstly elutes from the column, and was detected at ~9 min g⁻¹ after the gas mixture was introduced into the column. While Xe retains strongly in the column and was detected at a breakthrough time of ~15 min g⁻¹ for CPM-5, 24 min g⁻¹ for CPM-6 and 35 min g⁻¹ for Co²⁺-CPM-6, respectively. The Kr concentration at the column outlet exceeding the feed concentration indicates

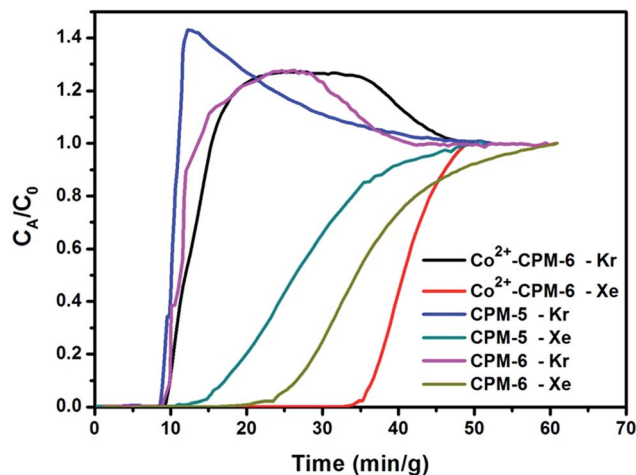


Fig. 7 The breakthrough curves of CPM-5, CPM-6 and Co²⁺-CPM-6 for a 20 : 80 Xe/Kr gas mixture at 298 K and 1 bar. The flow rate of He and Xe/Kr gas mixture is 5 ml min⁻¹.

a significantly stronger interaction for Xe molecules than Kr molecules with the adsorbents during the breakthrough process. The Xe/Kr selectivity of CPM-5, CPM-6 and Co²⁺-CPM-6 was calculated to be 6.2, 6.6 and 8.9 respectively, based on the adsorption capacity of the two gases, agreeable with the IAST selectivity derived from the single gas adsorption isotherms. These results indicate that after Co²⁺-exchange process, Co²⁺-CPM-6 has a remarkable higher Xe/Kr selectivity than CPM-5 and CPM-6 not only at equilibrium, but also under kinetic flow conditions. What's more, Co²⁺-CPM-6 exhibits less diffusion effect for Xe than CPM-5 and CPM-6 in the breakthrough process and would provide better Xe/Kr separation performance in the practical application. This could be due to the improvement of pore size and micropore volumes in the framework during the Co²⁺-exchange process. The high Xe capacity and Xe/Kr selectivity suggests Co²⁺-CPM-6 a promising candidate for Xe/Kr adsorptive separation.

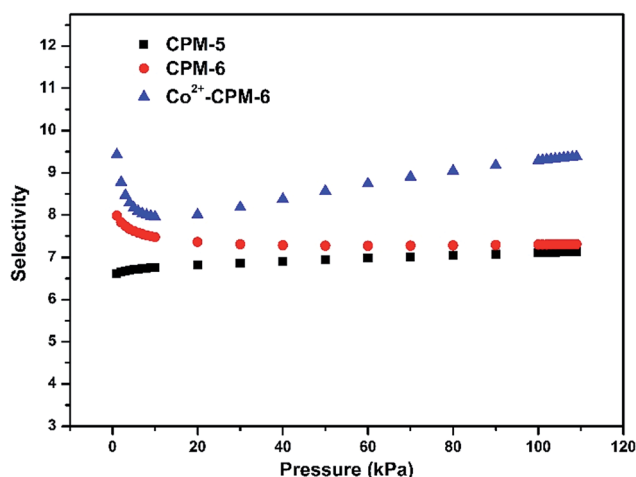


Fig. 6 IAST-predicted Xe/Kr selectivities of CPM-5, CPM-6 and Co²⁺-CPM-6 for 20/80 Xe/Kr gas mixture at 298 K.

Conclusions

In this work, we have investigated adsorptive Xe/Kr separation potential of three anionic porous indium–organic frameworks (CPM-5, CPM-6 and the Co²⁺-exchanged framework analogue Co²⁺-CPM-6). A new anionic In-MOF Co²⁺-CPM-6 with Co²⁺ ions in pore spaces has been obtained using a simple cation-exchange process and exhibits much higher Xe adsorption capacity and Xe/Kr selectivity than CPM-5 and CPM-6, verified by the single-component gas isotherms, IAST calculations and breakthrough experiments. Small and polarized Co²⁺ ions could increase the pore size or accessible micropore volumes and enhance the electric field within pore space, which could induce strong interaction with Xe and simultaneously reduce the affinity with Kr, suggesting the potential of Co²⁺-CPM-6 to be a promising candidate material for Xe/Kr adsorptive separation. The introduction of small and polarized metal ions such as Co²⁺ ions into an anionic MOF to tune its pore size or pore volume



and electric field across pore space may be a promising strategy to enhance Xe adsorption capacity and Xe/Kr separation performance of MOFs.

Conflicts of interest

Authors declare no conflicts of interest and have reached an agreement of the final version of manuscript.

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Notes and references

- 1 F. G. Kerry, *Industrial Gas Handbook: Gas Separation and Purification*, CRC Press, Boca Raton, Florida, 2007.
- 2 S. C. Cullen and E. G. Gross, *Science*, 1951, **113**, 580–582.
- 3 A. Turkevich, L. Winsberg, H. Flotow and R. M. Adams, *Proc. Natl. Acad. Sci. U. S. A.*, 1997, **94**, 7807–7810.
- 4 N. R. Soelberg, T. G. Garn, M. R. Greenhalgh, J. D. Law, R. Jubin, D. M. Strachan and P. K. Thallapally, *Sci. Technol. Nucl. Install.*, 2013, 72496.
- 5 J. Izumi, Waste Gas Treatment using Zeolites in Nuclear-Related Industries, in *Handbook of Zeolites Science and Technology*, Marcel Dekker, Inc., New York, 2003.
- 6 R. T. Yang, *Adsorbents: Fundamentals and Applications*, John Wiley & Sons, Inc., Hoboken, 2003.
- 7 C. G. Saxton, A. Kruth, M. Castro, P. A. Wright and R. F. Howe, *Microporous Mesoporous Mater.*, 2010, **129**, 68–73.
- 8 R. E. Bazan, M. Bastos-Neto, A. Moeller, F. Dreisbach and R. Staudt, *Adsorption*, 2011, **17**, 371–383.
- 9 X. Feng, Z. Zong, S. K. Elsaïdi, J. B. Jasinski, R. Krishna, P. K. Thallapally and M. A. Carreon, *J. Am. Chem. Soc.*, 2016, **138**, 9791–9794.
- 10 G. Férey, *Chem. Soc. Rev.*, 2008, **37**, 191–214.
- 11 S. Kitagawa, R. Kitaura and S. Noro, *Angew. Chem., Int. Ed.*, 2004, **43**, 2334–2375.
- 12 O. M. Yaghi, M. O'Keeffe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, *Nature*, 2003, **423**, 705–714.
- 13 D. M. D'Alessandro, B. Smit and J. R. Long, *Angew. Chem., Int. Ed.*, 2010, **49**, 6058–6082.
- 14 R. Matsuda, R. Kitaura, S. Kitagawa, Y. Kubota, R. V. Belosludov, T. C. Kobayashi, H. Sakamoto, T. Chiba, M. Tkata, Y. Kawazoe and Y. Mita, *Nature*, 2005, **436**, 238–241.
- 15 Z. Zhang, Z. Z. Yao, S. Xiang and B. Chen, *Energy Environ. Sci.*, 2014, **7**, 2868–2899.
- 16 S. Xiang, Y. He, Z. Zhang, H. Wu, W. Zhou, R. Krishna and B. Chen, *Nat. Commun.*, 2012, **3**, 954–960.
- 17 J. R. Li, R. J. Kuppler and H. C. Zhou, *Chem. Soc. Rev.*, 2009, **38**, 1477–1504.
- 18 P. Nugent, Y. Belmabkhout, S. D. Burd, A. J. Cairns, R. Luebke, K. Forrest, T. Pham, S. Ma, B. Space, L. Wojtas, M. Eddaoudi and M. J. Zaworotko, *Nature*, 2013, **495**, 80–84.
- 19 S. S. Xiong, Y. J. Gong, H. X. Wang, H. L. Wang, Q. Liu, M. Cu, X. L. Wang, B. L. Chen and Z. Y. Wang, *Chem. Commun.*, 2014, **50**, 12101–12104.
- 20 R. E. Morris and P. S. Wheatley, *Angew. Chem., Int. Ed.*, 2008, **47**, 4966–4981.
- 21 J. R. Li, Y. G. Ma, M. C. McCarthy, J. Sculley, J. M. Yu, H. K. Jeong, P. B. Balbuena and H. C. Zhou, *Coord. Chem. Rev.*, 2011, **255**, 1791–1823.
- 22 M. V. Parkes, C. L. Staiger, J. J. Perry IV, M. D. Allendorf and J. A. Greathouse, *Phys. Chem. Chem. Phys.*, 2013, **15**, 9093–9106.
- 23 T. Van Heest, S. L. Teich-McGoldrick, J. A. Greathouse, M. D. Allendorf and D. S. Sholl, *J. Phys. Chem. C*, 2012, **116**, 13183–13195.
- 24 Y. S. Bae, B. G. Hauser, Y. J. Colon, J. T. Hupp, O. K. Farha and R. Q. Snurr, *Microporous Mesoporous Mater.*, 2013, **169**, 176–179.
- 25 K. V. Lawler, Z. Hulvey and P. M. Forster, *Chem. Commun.*, 2013, **49**, 10959–10961.
- 26 C. A. Fernandez, J. Liu, P. K. Thallapally and D. M. Strachan, *J. Am. Chem. Soc.*, 2012, **134**, 9046–9049.
- 27 C. M. Simon, R. Mercado, S. K. Schnell, B. Smit and M. Haranczyk, *Chem. Mater.*, 2015, **27**, 4459–4475.
- 28 S. J. Lee, T. U. Yoon, A. R. Kim, S. Y. Kim, K. H. Cho, Y. K. Hwang, J. W. Yeon and Y. S. Bae, *J. Hazard. Mater.*, 2016, **320**, 513–520.
- 29 J. Liu, P. K. Thallapally and D. Strachan, *Langmuir*, 2012, **28**, 11584–11589.
- 30 M. H. Mohamed, S. K. Elsaïdi, T. Pham, K. A. Forrest, H. T. Schaefer, A. Hogan, L. Wojtas, W. Xu, B. Space, M. J. Zaworotko and P. K. Thallapally, *Angew. Chem., Int. Ed.*, 2016, **55**, 8285–8289.
- 31 Q. Wang, S. S. Xiong, Z. H. Xiang, S. M. Peng and D. P. Cao, *Sci. China: Chem.*, 2016, **59**, 643–650.
- 32 Q. Wang, H. Wang, S. M. Peng, X. Peng and D. P. Cao, *J. Phys. Chem. C*, 2014, **118**, 10221–10229.
- 33 S. K. Elsaïdi, M. H. Mohamed, C. M. Simon, E. Braun, T. Pham, K. A. Forrest, W. Xu, D. Banerjee, B. Space, M. J. Zaworotko and P. K. Thallapally, *Chem. Sci.*, 2017, **8**, 2373–2380.
- 34 R. S. Patil, D. Banerjee, C. M. Simon, J. L. Atwood and P. K. Thallapally, *Chem.–Eur. J.*, 2016, **22**, 12618–12623.
- 35 S. K. Ghose, Y. Li, A. Yakovenko, A. Yakovenko, E. Dooryhee, L. Ehm, L. E. Ecker, A. Dippel, G. J. Halder, D. M. Strachan and P. K. Thallapally, *J. Phys. Chem. Lett.*, 2015, **6**, 1790–1794.
- 36 J. Liu, C. A. Fernandez, P. F. Martin, P. K. Thallapally and D. M. Strachan, *Ind. Eng. Chem. Res.*, 2014, **53**, 12893–12899.
- 37 D. Banerjee, A. J. Cairns, J. Liu, R. K. Motkuri, S. K. Nune, C. A. Fernandez, R. Krishna, D. M. Strachan and P. K. Thallapally, *Acc. Chem. Res.*, 2015, **48**, 211–219.
- 38 H. Wang, K. Yao, Z. Zhang, J. Jagiello, Q. Gong, Y. Han and J. Li, *Chem. Sci.*, 2014, **5**, 620–624.
- 39 P. K. Thallapally, J. W. Grate and R. K. Motkuri, *Chem. Commun.*, 2012, **48**, 347–349.
- 40 J. Liu, D. M. Strachan and P. K. Thallapally, *Chem. Commun.*, 2014, **50**, 466–468.
- 41 L. Chen, P. S. Reiss, S. Y. Chong, D. Holden, K. E. Jelfs, T. Hasell, M. A. Little, A. Kewley, M. E. Briggs,



- A. Stephenson, K. M. Thomas, J. A. Armstrong, J. Bell, J. Busto, R. Boel, J. Liu, D. M. Strachan, P. K. Thallapally and A. I. Cooper, *Nat. Mater.*, 2014, **134**, 18892–18895.
- 42 S. S. Xiong, Q. Liu, Q. Wang, W. Li, Y. Tang, X. L. Wang, S. Hu and B. L. Chen, *J. Mater. Chem. A*, 2015, **3**, 10747–10752.
- 43 D. Banerjee, C. M. Simon, A. M. Plonka, R. K. Motkuri, J. Liu, X. Chen, B. Smit, J. B. Parise, M. Haranczyk and P. K. Thallapally, *Nat. Commun.*, 2016, **7**, 11831.
- 44 B. J. Sikora, C. E. Wilmer, M. L. Greenfield and R. Q. Snurr, *Chem. Sci.*, 2012, **3**, 2217–2223.
- 45 J. J. Perry, S. L. Teich-McGoldrick, S. T. Meek, J. A. Greathouse, M. Haranczyk and M. D. Allendorf, *J. Phys. Chem. C*, 2014, **118**, 11685–11698.
- 46 X. Chen, A. M. Plonka, D. Banerjee, R. Krishna, H. T. Schaef, S. Ghose, P. K. Thallapally and J. B. Parise, *J. Am. Chem. Soc.*, 2015, **137**, 7007–7010.
- 47 S. T. Meek, S. L. Teich-McGoldrick, J. J. Perry, J. A. Greathouse and M. D. Allendorf, *J. Phys. Chem. C*, 2012, **116**, 19765–19772.
- 48 S. K. Ghose, Y. Li, A. Yakovenko, E. Dooryhee, L. Ehm, L. Ecker, A.-C. Dippel, G. J. Halder, D. M. Strachan and P. K. Thallapally, *J. Phys. Chem. Lett.*, 2015, **6**, 1790–1794.
- 49 S.-T. Zheng, J. T. Bu, Y. F. Li, T. Wu, F. Zuo, P. Y. Feng and X. H. Bu, *J. Am. Chem. Soc.*, 2010, **132**, 17062–17064.
- 50 F. Nouar, J. Eckert, J. F. Eubank, P. Forster and M. Eddaoudi, *J. Am. Chem. Soc.*, 2009, **131**, 2864–2870.
- 51 T. Vazhappilly, T. K. Ghanty and B. N. Jagatap, *J. Phys. Chem. C*, 2016, **120**, 10968–10974.
- 52 S. J. Lee, K. C. Kim, T. U. Yoon, M. B. Kim and Y. S. Bae, *Microporous Mesoporous Mater.*, 2016, **236**, 284–291.
- 53 A. L. Myers and J. M. Prausnitz, *AIChE J.*, 1965, **11**, 121–127.
- 54 R. Babarao, Z. Q. Hu, J. W. Jiang, S. Chempath and S. I. Sandler, *Langmuir*, 2007, **23**, 659–666.
- 55 S. R. Challa, D. S. Sholl and J. K. Johnson, *J. Chem. Phys.*, 2002, **116**, 814–824.
- 56 L. Czepirski and J. Jagiello, *Chem. Eng. Sci.*, 1989, **44**, 797–801.

