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## Development of contact-killing non-leaching antimicrobial guanidyl-functionalized polymers *via* click chemistry

Kaimei Peng, ம<sup>a</sup> Tao Zou,<sup>a</sup> Wei Ding,<sup>a</sup> Ruonan Wang,<sup>a</sup> Jinshan Guo,<sup>b</sup> John Jay Round,<sup>b</sup> Weiping Tu,<sup>a</sup> Chao Liu<sup>b</sup> and Jianqing Hu<sup>\*a</sup>

An expedient and efficient method applied to synthesize click-suitable penta-substituted guanidines at room temperature is hereby described. The guanidine containing polyurethanes (PU-TMGs) were prepared *via* click reaction (copper catalyzed alkyne-azide [3 + 2] cycloaddition, CuAAC), which was performed either before or after polymerization to incorporate the guanidine. The guanidine-functionalized polymers exhibited strong contact-killing antibacterial activity against both Gram-negative and Gram-positive bacteria. There is around 99.9% killing of *S. aureus* and 98.0% killing of *E. coli* when TMG weight percentage content in PU-TMG is around 1%. Covalent conjugation of the guanidine groups to the polymers prevents leaching. The conjugation also reduces cytotoxicity of the materials and preserves long-term antimicrobial activity.

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## 1 Introduction

Microbes are ubiquitous in nature. Although some of them are beneficial to human beings, most of them are harmful, and some are even fatal. According to the list of major concerns about health from World Health Organization (WHO), various health problems are caused by micro-organisms such as influenza, cholera and sanitation. Microbes may contaminate water and sanitation services, and infect thousands of people. Therefore, some processes such as cleaning, disinfection and sterilization of wastewater and air are certainly essential. Although natural products and small molecules are used to prevent the spread and growth of microbes, the unpleasant side effects of releasing biocides into the environment, and the emerging antibiotic-resistant bacteria have become major concerns. Just in the European Union alone,<sup>1</sup> more than 25 000 people each year are contracted by antibiotic-resistant microbe, such as methicillin-resistant staphylococcus aureus (MRSA), one of several "superbugs" that are a global threat.<sup>2,3</sup>

The increasing prevalence of antibiotic resistance is a serious emerging challenge and has been a driving force for the development of new anti-microbial materials. In recent years, the development of cationic antimicrobial polymers has become a promising strategy. Different with other antimicrobial agents, the cationic polymers and the negatively charged microbial cell membranes undergo non-specific electrostatic interactions instead of specific receptor-protein interactions. Those types of interactions present minimal risk for the creation of mechanisms of bacterial resistance.<sup>4</sup> As a result of the critical attribute, quaternary ammonium polymers have widely used as cationic antimicrobial materials. However, the weak interactions between the quaternary ammonium and the phosphate groups of phospholipids within the cell membranes and the poor alkaline stability of the material at elevated temperatures are challenges that still need to be addressed.<sup>5</sup>

In recent years, guanidine-based polymers have attracted significant attention as powerful, and selective anti-microbial materials because the selective interactions occur at between guanidine groups and negatively charged bacterial cellmembrane instead of electrically neutral mammalian cell membranes, and nontoxic antimicrobial materials. Guanidine groups not only exist in natural antimicrobial products, such as antimicrobial peptides (AMPs),6 and play indispensable roles in asymmetric synthesis,7 but they also have been widely used in fuel cells,<sup>5,8,9</sup> antimicrobial,<sup>10-12</sup> biology,<sup>13,14</sup> medicine,<sup>15,16</sup> and catalysis study.<sup>17,18</sup> Guanidine groups have unique features that enable their antimicrobial capabilities.9 Guanidine groups are strong bases with a  $pK_a$  value of 13.8 for pentamethyl guanidine; The positive charges of the guanidine group are delocalized over three nitrogen atoms when guanidine is ionized, which contributes to a high degree of ionization in a wide pH range, a high thermal and basic stability, and a strong interaction with phosphate groups by forming hydrogen bonded ion pairs.<sup>19</sup> The bidendate binding between guanidine groups and phosphate groups<sup>20</sup> is much stronger than that between amine groups and phosphate groups,<sup>21</sup> resulting in a high bactericidal rate at fairly low antimicrobial concentrations of guanidine-based molecules or polymers. Importantly, the toxicity of charge-delocalized

<sup>&</sup>lt;sup>a</sup>School of Chemistry and Chemical Engineering, South China University of Technology, Guangzhou 510640, China. E-mail: glennhu@scut.edu.cn
<sup>b</sup>Aleo BME, Inc.200 Innovation Blvd. Suite 210A, State College, PA 16803, USA

cationic compounds, such as pyridinium and guanidine,<sup>22</sup> is much lower than that of charge-localized linear amine head group. Guanidine polymers possess both antimicrobial and low hemolytic characteristics.<sup>11</sup>

Although the applications of the polymers containing guanidine group located in the backbone chain such as polyhexamethylene guanidine oligomer (PHMG), polyhexamethylene biguanide (PHMB), poly(p-xylyleneguanidine) hydrochloride (PPXG) and guanidine and urea copolymer (PGUC) were reported,<sup>23-25</sup> the characteristic of high solubility in water restricted their wide application. Grafting guanidine to different polymer chain is a useful method to construct the nonleaching antimicrobial polymers and expands the application of guanidine-rich polymers. Moreover, the contact-killing mode is possible for the guanidine-rich polymer because the interactions of guanidine groups and negatively charged bacterial cellmembrane occur on the surface of cell. One method to prepare contact-killing polymer is physical mixing leaching microbicide and polymer,26,27 which cannot solve the sterilizing effect dropped over time and residual toxicity for environment.<sup>28</sup> Accordingly, the linking between polymer and guanidine groups is a promising method to construct non-leaching contact-killing antimicrobial guanidyl-functionalized polymers based on the properties of the guanidine-rich polymer.

At present, the most popular method for fabricating guanidyl-functionalized polymers is the quaternization of penta-substituted guanidines with bromobenzyl (or chlorobenzyl) containing polymers.5 However, this method has a limited reaction rate and efficiency, and lacks of orthogonality and an precise control over the degree and location of functionalization.<sup>29</sup> Furthermore, the stability of benzyl guanidines is poor as a result of nucleophilic attack of the benzylic cation by hydroxide8 or methoxyl17 ions and then through ylide formation and rearrangement reactions,30 which will release the guanidine groups and deactivate the guanidine-rich polymers quickly. Additionally, the current synthesis process of penta-substituted guanidines is complicated and not eco-friendly.31,32 It has been a challenge to develop a simple and highly efficient strategy for the preparation of penta-substituted guanidines and stable guanidyl-functionalized polymers.

The growing applications of polyurethanes stem from the ability to tailor their properties *via* step additional-polymerization between isocyanate and hydroxyl functionalities, which is the foundation of its compatibility with different components. Polyurethane allows for the introduction of functional molecules and the material exhibits excellent thermal and mechanical properties, all of which fuel the use of polyurethanes in adhesives,<sup>33</sup> sealants,<sup>34</sup> synthetic fibers,<sup>35</sup> foams,<sup>36</sup> coatings,<sup>37</sup> biomedical implants<sup>38</sup> and antimicrobial materials.<sup>39,40</sup>

Click chemistry, especially the favored copper-catalyzed azide-alkyne cycloaddition (CuAAC) process, is a robust, efficient and orthogonal method for the construction of functional molecules.<sup>41,42</sup> Recently, we have reported that click chemistry is a powerful tool for the polymer crosslinking and functionalization.<sup>43,44</sup> Here, a new facile approach was developed for the synthesis of a click-suitable penta-substituted guanidine (2-

propinyl-1,1,3,3-tetramethylguanidine, TMG-Al) *via* commercially available raw material at room temperature. A TMG-based antimicrobial polyurethane (PU-TMG) was prepared by either clicking TMG-Al onto azide diol (2,2-bis(azidomethyl)propane-1,3-diol, DAzD) to synthesize PU-TMG or by clicking TMG-Al onto the formed azide-containing polyurethane (PU-N<sub>3</sub>). Meanwhile, the physiochemical properties and the antimicrobial performance of the PU-TMG were investigated.

### 2 Materials and methods

#### 2.1 Materials

Trimethylolpropane (TMP), 1,4-butanediol (BDO), poly(propylene glycol) (M.W. = 1000, PPG 1000), and isophorone diisocyanate (IPDI) were purchased from J&K chemical Ltd. All other solvents and reagents for the monomer synthesis were purchased from Aldrich or Aladdin chemical Ltd. All other materials were purchased from J&K Acros Organics (Beijing, China). All chemicals were used as received and Milli-Q filtered water was used to prepare bacterial culture media. Polymerizations were performed in dry DMF purchased from Sigma. The two bacteria strains used were Gram-negative *Escherichia coli (E. coli*, ATCC 25922) and Gram-positive *Staphylococcus aureus (S. aureus*, ATCC 29213). Bacterial culture media were autoclaved before using to ensure sterility.

#### 2.2 Methods

2.2.1 Synthesis. 2,2-Bis(azidomethyl)propane-1,3-diol (DAzD) was synthesized as described.43 Briefly, 2.2bis(bromomethyl)propane-1,3-diol (98%, 10.4 g, 40 mmol) was dissolved in dimethyl sulfoxide (DMSO, 40 mL) in a 100 mL 3neck flask equipped with a reflux condenser. Sodium azide (6.5 g, 100 mmol) was added under nitrogen. The suspension was then heated to 110 °C and stirred for 20 hours. After being cooled to room temperature, 30 mL of water was added and the mixture was transferred to a 500 mL separatory funnel and subsequently extracted with ethyl acetate (90 mL  $\times$  3). The combined organic phase was washed by saturated brine (60 mL  $\times$  3) and then dried with sodium sulfate. After rotary evaporation, the final product (yellow liquid) was put under vacuum for at least 3 days before use.

1,1,3,3-Tetramethyl-2-(prop-2-yn-1-yl)guanidine (TMG-Al) was synthesized according to the following procedure. To a 100 mL flask charged with TMG (20 g, 0.17 mol) and 25 mL ethyl acetate, after being cooled by ice bath, propargyl bromide (7.14 g, 0.06 mol) was slowly added with vigorous stirring. Then, stirring was continued for another 16–24 hours at room temperature, and a large amount of precipitate occurred. After completion of the reaction, the solid was filtered. After aqueous workup, extraction with dichloromethane (DCM), wash with brine, and concentration of the reaction mixture, a brown liquid (5.51 g) was obtained in 60% yield.

Diguanidine diol (TMG-OH) was synthesized *via* CuAAC as follows. A 100 mL flask charged with TMG-Al (1.83 g, 11.84 mmol), DAzD (1.00 g, 5.38 mmol) and 30 mL ethyl acetate was degassed by six freeze-pump-thaw cycles. Subsequently, the

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prepared catalyst solution, 5 mL dimethylformamide (DMF) solution containing 0.2 g CuBr and 0.36 mL N,N,N',N'',N''-pentamethyl diethylenetriamine (PMDETA), was then transferred the reaction mixture under nitrogen atmosphere. The reaction is completed in 1 hour at room temperature and the viscous solid precipitated. The liquid was removed and, the final solid product was obtained after vacuum dry for at least 48 hours. Yield: 2.44 g (92%).

The click-suitable azide functional polyurethane (PU-N<sub>3</sub>) was prepared according to the following procedure. Firstly, IPDI (30 g, 135 mmol) and PPG 1000 (27.5 g, 27.5 mmol) were charged into a 250 mL dried four-necked flask equipped with a mechanical stirrer, a thermometer, a condenser, and a nitrogen in/outlet. After stirring for 10 minutes to give a uniform mixture, the temperature was increased to 80 °C and maintained reaction for 2 hours. Then, 140 mL DMF, BDO (4.68 g, 52 mmol), TMP (3 g, 22 mmol) and DAzD (2.75 g, 15 mmol) were added into the flask and the reaction was kept for another 6–7 hours. PU-N<sub>3</sub> in DMF (solid content: 34%) was prepared.

Route 1. Polyurethane with side-chain tetramethylgermanium (PU-TMG) was prepared according to the following procedure. The click reaction was performed between PU-N<sub>3</sub> and TMG-Al as follows: 15 mL water/DMF mixture ( $V_{\text{DMF}}$  :  $V_{\text{H}_2\text{O}}$ = 9:1), and a certain amount of TMG-Al (0.030 g for PU-TMG 1, 0.052 g for PU-TMG 2, 0.109 g for PU-TMG 3, and 0.297 g for PU-TMG 4) were added to 15 g solution of PU-N<sub>3</sub> in DMF at room temperature under stirring, followed by the addition of 0.01 g CuSO<sub>4</sub>·5H<sub>2</sub>O in 1 mL water–DMF mixed solvent ( $V_{\text{DMF}}$  :  $V_{\text{H}_{2}\text{O}}$  = 9:1). The mixture was degassed for 20 minutes with nitrogen and 0.03 g sodium ascorbate in 1 mL water-DMF mixed solvent was subsequently added in, and then the click reaction was initiated under nitrogen atmosphere. The mixture was stirred for additional 1 hour, then 10 g of the obtained mixture was poured into a 4 cm diameter-teflon disk and dried at 75 °C for 24 hours. The PU-TMGs were precipitated from water and dried. The yields of the final polymer PU-TMGs remained above 98%.

*Route 2.* Tetramethylgermanium containing polyurethane (PU-TMG 1'-4') was prepared according to the following procedure. IPDI (30 g, 135 mmol) and PPG 1000 (27.5 g, 27.5 mmol) were charged into a 250 mL dried four-necked flask equipped with a mechanical stirrer, a thermometer, a condenser, and a nitrogen in/outlet. After stirring for 10 minutes to give a uniform mixture, the temperature was increased to 80 °C, and the reaction was maintained for 2 hours. Then, 140 mL DMF, TMP (3 g, 22 mmol), a certain amount of TMG-Al and BDO were added into the flask and the reaction was kept for another 6–7 hours. Finally, the PU-TMG 1'–4' have the same TMG content as the PU-TMG 1–4 were obtained. The PU-TMGs were precipitated from water and dried. The yields of the final polymer PU-TMGs remained above 98%.

**2.2.2** Spectroscopic characterization. <sup>1</sup>H and <sup>13</sup>C NMR spectra were recorded on Bruker AVANCE III HD 400 spectrometer at room temperature, using  $CDCl_3$  or DMSO-d<sub>6</sub> as solvent. FT-IR spectrum was recorded on Bruker VERTEX 70 in the range of 4000–600 cm<sup>-1</sup> using the KBr disc method at room temperature.

**2.2.3 Gel permeation chromatography (GPC).** The number average molecular weights  $(M_n)$ , the weight average molecular weights  $(M_w)$ , and polydispersity indexes (PDI) of the PU-TMGs were evaluated by gel permeation chromatography (GPC) using a Thermo Scientific chromatograph equipped with an isocratic Dionex UltiMate 3000 pump and a RefractoMax 521 refractive index detector. The system was operated using tetrahydrofuran as the eluent at a flow rate of 1 mL min<sup>-1</sup> and calibrated using polystyrene standards.

**2.2.4 Water absorption and contact angle.** The films of specific dimensions  $(3 \text{ cm} \times 3 \text{ cm} \times 0.6 \text{ mm})$  were immersed in deionized water at room temperature for 72 hours. After being wiped with tissue paper, the membrane was quickly weighed on a microbalance. Water absorption percentage of samples can be calculated by

Water absorption percentage = 
$$\frac{w_2 - w_1}{w_1} \times 100\%$$
 (1)

where  $w_2$  and  $w_1$  are the masses of water-swollen and dry films, respectively.

The static contact angles of films were conducted on a PocketGoniometer PGX+ contact angle analyser. The average contact angle was obtained from five tests of each sample.

2.2.5 Antimicrobial activity determination. The antibacterial activity analysis of different films was performed on both Gram-positive S. aureus (ATCC 29213) and Gram-negative bacterial E. coli (ATCC 25922) strains using the so-called antimicrobial drop-test.45,46 Bacteria from a deep frozen aliquot  $(-20 \,^{\circ}\text{C})$  were plated out on an agar plate and incubated overnight at 37 °C for 24 hours, and then a single colony was picked and cultured in culture medium at 37 °C for 12 hours. Afterwards, the bacteria suspensions were diluted to about 10<sup>5</sup> colony forming units (CFUs) per mL for the dropmethod antimicrobial experiments. The samples (20  $\times$  20  $\times$ 0.5 mm) were placed in the sterilized petri dishes, and then 100 µL of 10<sup>5</sup> colony forming units (CFUs) per mL bacteria suspensions was added dropwise onto the surface of each samples, pure PU was used as a control. All the samples were laid at 37 °C for 8 hours, washed with 10 mL sterile doubledistilled water. For the washing process, 1 mL sterile doubledistilled water was added dropwise onto the film surface each time, and the surface was washed with the help of pipette. Then 100 µL of bacteria suspension was dispersed on the nutrient agar plate. The numbers of surviving bacteria on the agar plates were counted after incubation for 24 hours at 37 °C. The above experiment was repeated at least three times and the polymer samples were washed with ethanol and sterile double-distilled water and dried at 75 °C before the experiment.

The disk diffusion of different samples was conducted according to the following procedure. The 24 hours cultured active bacterial suspensions were spreaded on the surface of the solidified agar media. Polymer films with  $10 \times 10 \times 0.5$  mm were adhered closely to the medium and the architecture was incubated at 37 °C for 8 hours. The polymer samples were washed with ethanol and sterile double-distilled water and dried at 75 °C before the experiment.

**2.2.6 Thermogravimetric analysis (TGA).** Thermogravimetric analysis (TGA) was conducted using NETZSCH, TG209 under  $N_2$  flow. The samples were heated from room temperature to 600 °C with a heating rate of 10 °C min<sup>-1</sup>.

**2.2.7 Differential scanning calorimetry (DSC).** DSC tests were conducted using a Netzsch DSC204 differential scanning calorimeter. The sample was first heated from room temperature to  $150 \,^{\circ}$ C at  $10 \,^{\circ}$ C min<sup>-1</sup> and held in the melting state for 2 minutes to erase the thermal history. This process was followed by quenching to  $-80 \,^{\circ}$ C at  $10 \,^{\circ}$ C min<sup>-1</sup> and held for 8 minutes to reach equilibrium. In the second run, the sample was heated from -80 to  $150 \,^{\circ}$ C at  $10 \,^{\circ}$ C min<sup>-1</sup> again to investigate the thermal property. The DSC peak separation technique was adopted using PEAK SEPARATION software provided by Netzsch Corp. with the DSC204 equipment.

### 3 Results and discussion

#### 3.1 Monomers synthesis

The basic strategies employed for the synthesis of DAzD, TMG-Al and TMG-OH are based on the nucleophilic substitution. DAzD was synthesized in one step from the 2,2-bis(bromomethyl) propane-1,3-diol in Fig. 1A and the structure was confirmed by  $^{1}$ H/ $^{13}$ C NMR and FT-IR spectra shown in Fig. 2A, C and E. The appearance of the characteristic infrared absorption peak of azide group was observed at 2104 cm<sup>-1</sup> of FTIR spectrum (Fig. 2E), and the left shift of the chemical shift of the protons on  $CH_2$  link to Br groups of the starting material at around 3.20 ppm to 3.70 ppm of the protons on  $CH_2$  link to azide groups of DAzD (<sup>1</sup>H NMR, Fig. 2A).<sup>43,44</sup> Generally, pentasubstituted guanidine was synthesized in two steps. Briefly, the intermediate product, Vilsmeier salt, was obtained by tetraalkyl urea and phosgene, and then the active intermediate reacted with primary amine.<sup>47</sup> Subsequently, Zhang<sup>48</sup> used the phosphorus oxychloride or thionyl chloride instead of hypertoxic phosgene, but the operation was also somewhat complicated due to the sensitivity of phosphorus oxychloride or thionyl chloride to water.

Herein, we firstly introduced a facile method to fabricate click-suitable penta-substituted guanidine by substitution reaction at room temperature in one step, and tetramethyl guanidine (TMG) is both reactive raw material and acid removal agent of side reaction products (Fig. 1B). The successful synthesis of TMG-Al was confirmed by the appearance of the characteristic infrared absorption peak of alkyne group at 2112 cm<sup>-1</sup>, the <sup>13</sup>C-NMR spectrum characteristic peak of central carbon atom in the guanidyl group (-N=C-) at around 162 ppm and the <sup>1</sup>H NMR peaks of protons on  $-CH_2-C \equiv CH$  ( $\delta$  2.12 ppm, s), -N(CH<sub>3</sub>)<sub>2</sub> ( $\delta$  2.67 ppm, s), -N(CH<sub>3</sub>)<sub>2</sub> ( $\delta$  2.75 ppm, s) and -CH<sub>2</sub>- $C \equiv C (\delta 3.92 \text{ ppm, s})$  in Fig. 2B, D and F. For comparison, and to meet various application requirements, hydroxyl-functionalized tetramethyl guanidine (TMG-OH) was prepared in accordance with Fig. 1C and its chemical structural characterizations are shown in Fig. 2G and H.

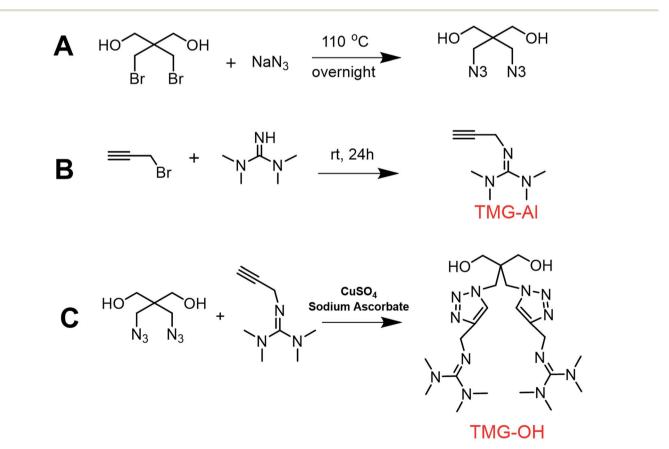


Fig. 1 Synthetic routes for the monomers.

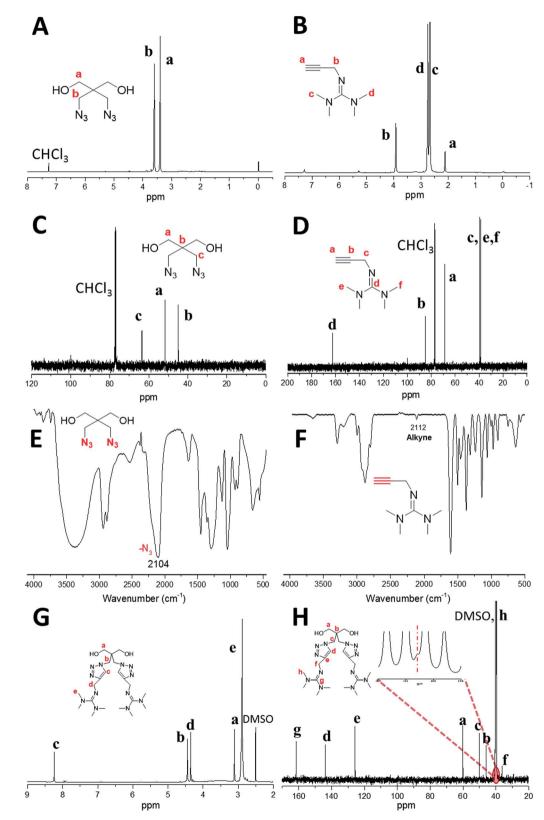


Fig. 2 <sup>1</sup>H NMR, <sup>13</sup>C NMR and FT-IR spectra of DAzD (A, C and E), TMG-Al (B, D and F) and <sup>1</sup>H NMR, <sup>13</sup>C NMR spectra of TMG-OH (G and H) in CDCl<sub>3</sub> or DMSO-*d*<sub>6</sub>.

#### 3.2 Polymer design and synthesis

Herein, we showed two synthetic strategies to fabricate the TMG functionalized polyurethanes. As illustrated in Fig. 3, firstly, IPDI–PPG based polyurethane prepolymer was obtained by step growth polymerization. Two synthetic strategies were then used to create guanidine-functionalized polymers. The first strategy was post-polymerization modification of azide functional polyurethanes, which was accomplished by adding an azide-functional diol (DAzD) directly.<sup>49</sup> In contrast, the second strategy is direct polymerization of TMG functional diol that obtained by click reaction of azide diol (DAzD) and alkyne functional TMG, which is a wise choice for the water dispersible polyurethane or polyester dispersions.

The structure of the click-suitable polymer (PU-N<sub>3</sub>) and polyurethane with side-chain tetramethylgermanium (PU-TMG) was confirmed by NMR and IR spectra shown in Fig. 4 and 5. Generally, the characteristic peak of imine group -C=N- is at 1675 cm<sup>-1</sup>,<sup>50</sup> but its absorption band overlapped with that of urethane groups -NHCOO-, so no obvious peak emerged in Fig. 4 after the introduction of TMG to PU-N<sub>3</sub>. As shown in Fig. 4, the characteristic peak of azide groups on PU-N<sub>3</sub> at 2106 cm<sup>-1</sup> significantly decreased after "click" reaction (CuAAC) between PU-N<sub>3</sub> and TMG-Al. Compared with pure PU, in the NMR spectra of PU-TMG polymers shown in Fig. 5, a new peaks appeared at around  $\delta$  2.88–2.81 ppm, which could be assigned to the chemical shifts of the protons on the methyl groups,  $-N(CH_3)_2$ , in guanidines,<sup>9</sup> and its strength increased with the

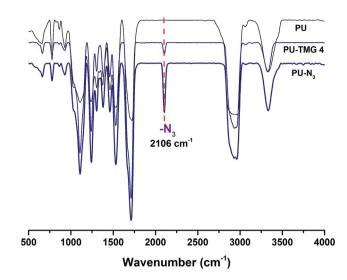


Fig. 4 FT-IR spectra of polyurethane (PU), click-suitable polyurethane (PU– $N_3$ ) and polyurethane with side-chain tetramethylgermanium (PU-TMG 4).

increase of content of TMG in the polymers. The PU-TMGs were subjected to GPC analysis. The number average molecular weights ( $M_n$ ) and the polydispersity indexes (PDI) of the PU-TMG 2 (PU-TMG 2') were 2.203 × 10<sup>4</sup> g mol<sup>-1</sup> (2.115 × 10<sup>4</sup> g mol<sup>-1</sup>) and 2.50 (2.74), respectively. It showed that there had no significant difference on the number average molecular weights

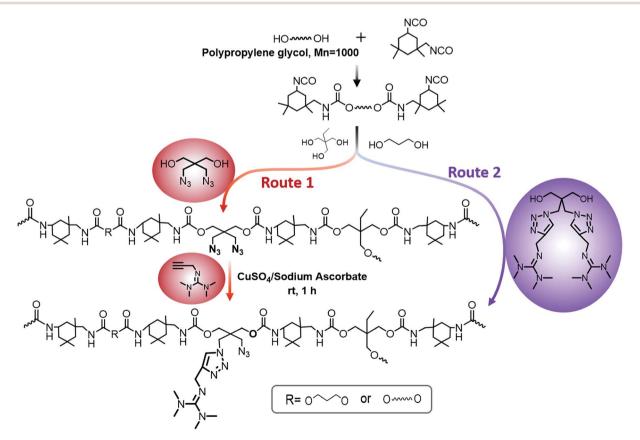


Fig. 3 Two synthetic strategies for the polymers.

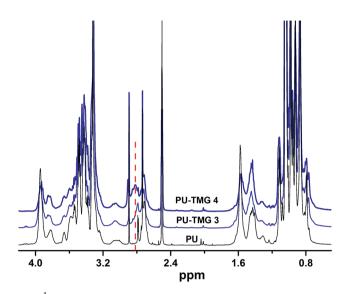


Fig. 5  $^{1}$ H NMR spectra of polyurethane (PU) and polyurethanes with side-chain tetramethylgermanium (PU-TMG 3 and PU-TMG 4) in DMSO- $d_{6}$ .

 $(M_n)$  and the polydispersity indexes (PDI) of PU-TMG prepared by route 1 or 2. The results of FTIR, NMR and GPC indicate that the guanidine-functionalized polyurethanes have been synthesized successfully.

#### 3.3 Thermal properties

Thermal analysis, including DSC and TGA, was used to investigate the thermal properties of PU, PU-N<sub>3</sub> and PU-TMG. The glass transition temperatures ( $T_{\rm g}$ s) for PU, PU-N<sub>3</sub>, PU-TMG 2, PU-TMG 3 and PU-TMG 4 were 28.0, 25.1, 25.8, 33.6, and 35.1 °C respectively. As shown in Fig. 6, Tg of PU-TMG gradually increased from 25.8 °C to 35.1 °C with the increasing of TMG content, which may be caused by the formation of rigid triazole rings by click reaction (CuAAC)<sup>43,44</sup> and the introduction of rigid

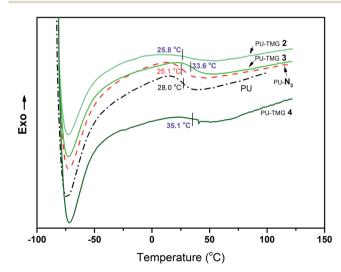


Fig. 6 DSC curves of PU, PU-N<sub>3</sub>, PU-TMG 2 (wt% = 1.0%), PU-TMG 3 (wt% = 2.1%), and PU-TMG 4 (wt% = 5.5%).

TMG groups. The decomposition temperature of PU-TMG polymers are around 200 °C.

Derivative thermogravimetric analysis (DTG) were also performed, which can give important information of polymer system, such as each step of weight loss corresponding to particular degradation process. As described in references, 51,52 PU decomposition occurs in two53 or three steps.54,55 In this particular work, all samples underwent a two-step thermal degradation. The first stage is the disintegration of PU polymer chains because the C-NH is the weakest bond in PU polymer and the activation energy is only about 98 kJ mol<sup>-1</sup>.<sup>51</sup> Generally, the urethane and urea bond start to decompose and dissociate into polyol, amine, isocyanate, olefin and carbon dioxide at 200 °C. Subsequently, as shown in Fig. 7B, the degradation is split into two peaks. The first peak is located near about 325 °C for pure PU, at which the amount of weight loss is  $\sim$ 56% and well correlated with the mass ration of hard segment in polyurethane (see Fig. 7A). At this point, all the urethane and urea bonds were mainly broken. The second stage was found to correlate with the degradation of soft segments. In this stage,

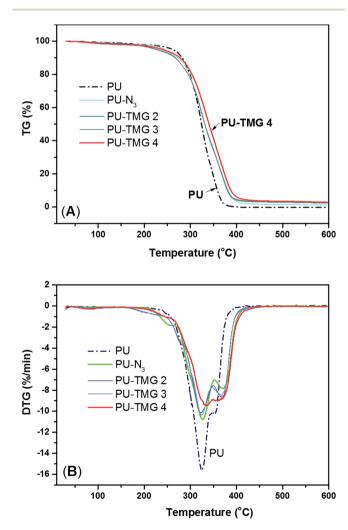


Fig. 7 TGA (A) and DTG (B) curves of PU, PU-N<sub>3</sub>, PU-TMG 2 (wt% = 1.0%), PU-TMG 3 (wt% = 2.1%), and PU-TMG 4 (wt% = 5.5%) under nitrogen.

the ether bond (C–O), whose bond energy is 326 kJ mol<sup>-1</sup>, began to break down. In addition, the rate of thermal degradation of PU-N<sub>3</sub> and PU-TMG was markedly slower than that of pure PU at first peak, which was attributed to the substitution of BDO with DAzD and the conjugation of TMG groups. As shown in in Fig. 7A, with the converting azide group into the thermal stable triazole ring formed by CuAAC, the thermal stability of PU-TMG became better, whose decomposition temperature reach up to 450 °C.<sup>56</sup> The result is in line with the increase of  $T_g$  of PU-TMG in comparison to PU-N<sub>3</sub>. Hence, CuAAC can not only serves as the perfect tool for the introduction of functional molecules onto polymers, but also improves the thermal stability and increases the  $T_g$  by the thermal stable and rigid triazole rings.

#### 3.4 Water-resistance

Water absorption is known to have a profound effect on films' antimicrobial durability of and mechanical properties.<sup>57</sup> Appropriate water absorption can help hydrophilic groups, such as guanidine group, to rearrange and aggregate to the surface of the films, which is beneficial to the antimicrobial activity of the antimicrobial polymer with a contact-killing mechanism. However, high levels of water absorption can result in the

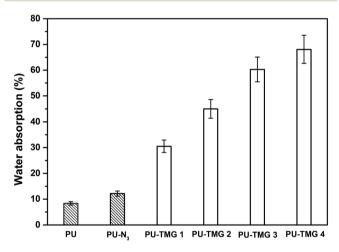


Fig. 8 Water absorption properties of PU, PU-N<sub>3</sub>, PU-TMG 1 (wt% = 0.6%), PU-TMG 2 (wt% = 1.0%), PU-TMG 3 (wt% = 2.1%), and PU-TMG 4 (wt% = 5.5%) at room temperature for 72 hours.

Table 1	Antimicrobial	activities	of	PU-TMG

detachment of film from the substrate, reduction of mechanical properties and leaching of a certain amount of antimicrobial polymer. Normally, the water absorption of quaternary ammonium or guanidine-rich polymers is strongly dependent on the density of quaternary ammonium or guanidine groups in polymer chains. The water absorption was determined by the weight percentage of water absorbed by the films when immersed into water at room temperature for 72 hours, in comparison to the dry samples weight. As shown in Fig. 8, the water absorption of PU-N<sub>3</sub> is larger than that of pure PU because the azide group is a more hydrophilic group compared with alkyl group. As expected, the water absorption of PU-TMG increased with the rise of the content of TMG in polymer. The water absorptions of PU, PU-N3, PU-TMG 1, PU-TMG 2 were 8.3%, 12.2%, 30.5% and 45.0%, respectively. It is consistent with static contact angle of PU (72.1°), PU-N<sub>3</sub> (66.1°), PU-TMG 1 (64.3°), and PU-TMG 2 (61.6°). Furthermore, no obvious film white phenomenon for the PU-TMG films was observed. These results showed that the water-resistance properties of PU-TMGs are acceptable.

#### 3.5 Antimicrobial activity

To evaluate antimicrobial properties of PU-TMG films, both Gram-positive *S. aureus* and Gram-negative *E. coli* were tested using the antibacterial "drop test" method.<sup>45,46</sup> PU was used as control. The percentage of bacteria kill was calculated by eqn (2) as follow:

Percentage of bacteria kill = 
$$\frac{A - B}{A} \times 100\%$$
 (2)

Here A = CFUs after control (PU) surface contact, B = CFUs after PU-TMG surface contact.

As shown in Table 1 and Fig. 11, the PU-TMG films showed a bacteria kill rate >93% against *S. aureus* and >60% against *E. coli*. When the TMG mass content was as low as 0.6% (PU-TMG 1), there was still a percentage of bacteria kill rate >90% against *S. aureus*. The results demonstrated that TMG containing polymer exhibited excellent antimicrobial characteristic. As shown in Fig. 9 and 10, it can be seen that increasing TMG mass content from 0.6% to 2.1% significantly improved the bacteria contactkilling activities of PU surfaces for either strain of bacteria and

Surface		% Bacteria killed					
	% TMG content in polymer	S. aureus (Gram-positive)		E. coli (Gram-negative)			
		Route 1	Route 2	Route 1	Route 2		
PU	0	0	0	0	0		
PU-TMG 1 (1')	0.6	$96.9\pm1$	$93.2\pm4$	$62.7\pm3$	$60.2\pm4$		
PU-TMG 2 $(2')$	1.0	99.9	$97.4\pm2$	$98.0\pm1$	$97.2\pm0.6$		
PU-TMG 3 (3')	2.1	99.9	99.9	99.9	99.9		
PU-TMG 4 $(4')$	5.5	99.9	99.9	99.9	99.9		

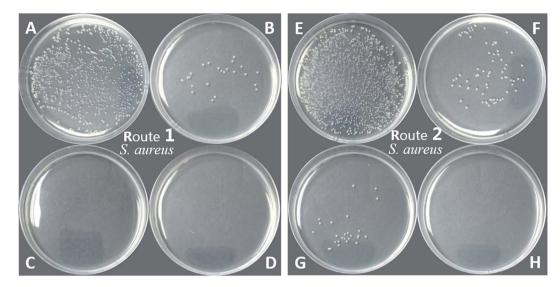


Fig. 9 Anti-microbial activity of PU (blank, A and E), PU-TMG 1 (B), PU-TMG 2 (C), PU-TMG 3 (D) (prepared by route 1), and PU-TMG 1' (F), PU-TMG 2' (G), PU-TMG 3' (H) (prepared by route 2) for *Staphylococcus aureus*.

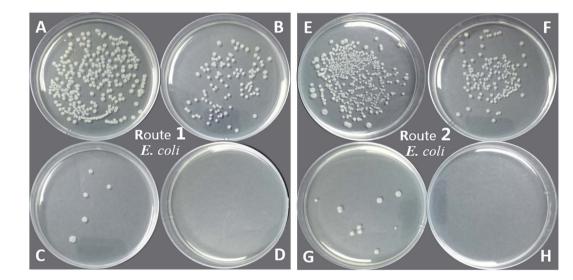


Fig. 10 Anti-microbial activity of PU (blank, A and E), PU-TMG 1 (B), PU-TMG 2 (C), PU-TMG 3 (D) (prepared by route 1), and PU-TMG 1' (F), PU-TMG 2' (G), PU-TMG 3' (H) (prepared by route 2) for Escherichia Coli.

the 1.0% TMG content in polymer is sufficient enough to inhibit *S. aureus* or *E. coli*. It is in accordance with the mechanism that the key determining factor in the antibacterial efficacy is the density of positive charges on the surface.<sup>28,45,58,59</sup> From Table 1, Fig. 8–11, we also observed that PU-TMG films have better antimicrobial activity against *S. aureus* than against *E. coli*, which is in line with the results of Saif *et al.* on quaternary ammonium antibacterial hybrid coating.<sup>60</sup> Until now, different mechanisms have been put forward for interpreting the action of killing bacteria<sup>28,61-63</sup> and the differentiated antimicrobial activities toward *E. coli* or *S. aureus* may stem from the different membrane structures of Gram-positive and Gram-negative bacteria.<sup>45</sup>

On the other hand, PU-TMG polymers synthesized through different ways, route 1 and route 2, exhibited almost the same antimicrobial activities (Fig. 9 and 10).

The disk diffusion of PU-TMG and the physical mixture of pure PU and TMG-OH films were carried out on both Gram positive and Gram negative bacterial strains. PU-TMG and physical mixed films (PU and TMG-OH) with  $10 \times 10 \times 0.5$  mm were adhered on the Luria Agar medium and the challenged bacteria was incubated at 37 °C for 8 hours. The physical mixed films showed a noticeable inhibition zone as shown in Fig. 12B and D, which means the small molecule TMG-OH can be leached from the mixed film due to non-covalently attaching between the polymer and functional small molecules. On the contrary, the surface of PU-TMG 2 sample was clean and there was no zones of inhibition around PU-TMG film in Fig. 12A and C, which demonstrates that no molecules were leached from PU-TMG film into the environment to kill the bacteria. Instead, the PU-TMG surfaces is capable of killing microorganisms on

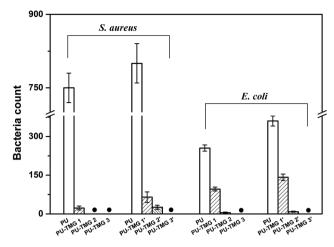
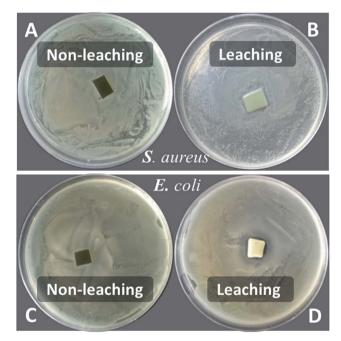


Fig. 11 The surviving bacteria (*S. aureus* or *E. coli*) counts on the agar after contacting with the PU, PU-TMG1(1'), PU-TMG2(2'), or PU-TMG 3 (3').



**Fig. 12** Disk diffusion of PU-TMG 2 (A and C) and the mixed film (PU + TMG-OH, B and D) for *Staphylococcus aureus* and *Escherichia coli*.

contact. This again proves that click chemistry is an excellent and practical tool to conjugate functional molecules, both before and after polymerization.

## 4 Conclusions

In summary, as an feasible, efficient chemical reaction that can be efficiently performed under mild conditions in different reaction media, "click" reaction (copper(i)-catalyzed alkyneazide cycloaddition, CuAAC) was employed to introduce antimicrobial tetramethyl guanidine (TMG) onto polymers before or after polymerization. It is clear that the click-suitable pentasubstituted guanidine (TMG-Al) represents an interesting and pretty important new class of molecules. Click reactions among various polymer backbones that contain azide groups and TMG-Al is an extremely efficient method to fabricate antimicrobial materials. The obtained guanidine functionalized polymers exhibited excellent antimicrobial property. Most importantly, unlike polymers mixed with antimicrobial agents/drugs, polymers with covalently bound guanidine groups are contactkilling but non-releasing antimicrobial material. The efficient, robust and orthogonal approach of click chemistry shown here will enable precise control and fine-tune of parameters such as charge-density and functional properties, fabricating a wide variety of well-defined functional macromolecules ultimately leading to a functionally versatile class of all-in-one materials.

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## References

- 1 O. Cars, A. Hedin and A. Heddini, *Drug Resist. Updates*, 2011, 14, 68–69.
- 2 E. Y. Furuya and F. D. Lowy, Nat. Rev. Microbiol., 2006, 4, 36–45.
- 3 H. F. Chambers and F. R. DeLeo, *Nat. Rev. Microbiol.*, 2009, 7, 629–641.
- 4 K. Kuroda and G. A. Caputo, *Wiley Interdiscip. Rev.: Nanomed. Nanobiotechnol.*, 2013, 5, 49–66.
- 5 L. Liu, Q. Li, J. Dai, H. Wang, B. Jin and R. Bai, *J. Membr. Sci.*, 2014, **453**, 52–60.
- 6 A. T. Yeung, S. L. Gellatly and R. E. Hancock, *Cell. Mol. Life Sci.*, 2011, **68**, 2161–2176.
- 7 M. P. Coles, *Chem. Commun.*, 2009, 3659–3676, DOI: 10.1039/ b901940e.
- 8 D. S. Kim, A. Labouriau, M. D. Guiver and Y. S. Kim, *Chem. Mater.*, 2011, 23, 3795–3797.
- 9 J. Wang, S. Li and S. Zhang, *Macromolecules*, 2010, **43**, 3890–3896.
- 10 K. E. S. Locock, T. D. Michl, N. Stevens, J. D. Hayball, K. Vasilev, A. Postma, H. J. Griesser, L. Meagher and M. Haeussler, ACS Macro Lett., 2014, 3, 319–323.
- 11 G. J. Gabriel, A. E. Madkour, J. M. Dabkowski, C. F. Nelson, K. Nusslein and G. N. Tew, *Biomacromolecules*, 2008, 9, 2980– 2983.
- 12 J. Budhathoki-Uprety, L. Peng, C. Melander and B. M. Novak, ACS Macro Lett., 2012, 1, 370–374.
- 13 E. M. Kolonko and L. L. Kiessling, J. Am. Chem. Soc., 2008, 130, 5626–5627.
- 14 C. B. Cooley, B. M. Trantow, F. Nederberg, M. K. Kiesewetter, J. L. Hedrick, R. M. Waymouth and P. A. Wender, J. Am. Chem. Soc., 2009, 131, 16401–16403.
- 15 R. J. Ono, A. L. Z. Lee, W. Chin, W. S. Goh, A. Y. L. Lee, Y. Y. Yang and J. L. Hedrick, *ACS Macro Lett.*, 2015, 4, 886– 891.

- 16 E. A. Dubikovskaya, S. H. Thorne, T. H. Pillow, C. H. Contag and P. A. Wender, *Proc. Natl. Acad. Sci. U. S. A.*, 2008, **105**, 12128–12133.
- 17 U. Schuchardt, R. M. Vargas and G. Gelbard, *J. Mol. Catal. A: Chem.*, 1996, **109**, 37–44.
- 18 C. Savelli and R. Salvio, Chemistry, 2015, 21, 5856-5863.
- 19 M. Onda, K. Yoshihara, H. Koyano, K. Ariga and T. Kunitake, *J. Am. Chem. Soc.*, 1996, **118**, 8524–8530.
- 20 J. B. Rothbard, E. Kreider, C. L. VanDeusen, L. Wright,
   B. L. Wylie and P. A. Wender, *J. Med. Chem.*, 2002, 45, 3612–3618.
- 21 K. E. Locock, T. D. Michl, J. D. Valentin, K. Vasilev, J. D. Hayball, Y. Qu, A. Traven, H. J. Griesser, L. Meagher and M. Haeussler, *Biomacromolecules*, 2013, 14, 4021–4031.
- 22 H. Lv, S. Zhang, B. Wang, S. Cui and J. Yan, *J. Controlled Release*, 2006, **114**, 100–109.
- 23 D. Wei, Q. Ma, Y. Guan, F. Hu, A. Zheng, X. Zhang, Z. Teng and H. Jiang, *Mater. Sci. Eng.*, *C*, 2009, **29**, 1776–1780.
- 24 M. Lebrini, F. Bentiss, N.-E. Chihib, C. Jama, J. P. Hornez and M. Lagrenée, *Corros. Sci.*, 2008, **50**, 2914–2918.
- 25 W. Li, H. Wang, Y. Ding, E. C. Scheithauer, O.-M. Goudouri, A. Grünewald, R. Detsch, S. Agarwal and A. R. Boccaccini, *J. Mater. Chem. B*, 2015, 3, 3367–3378.
- 26 S. D. Worley and G. Sun, Trends Polym. Sci., 1996, 4, 364–370.
- 27 W. K. Son, J. H. Youk and W. H. Park, *Carbohydr. Polym.*, 2006, **65**, 430–434.
- 28 J. Y. Huang, R. R. Koepsel, H. Murata, W. Wu, S. B. Lee, T. Kowalewski, A. J. Russell and K. Matyjaszewski, *Langmuir*, 2008, 24, 6785–6795.
- 29 Q. Zhang, S. Li and S. Zhang, Chem. Commun., 2010, 46, 7495-7497.
- 30 S. Chempath, J. M. Boncella, L. R. Pratt, N. Henson and B. S. Pivovar, J. Phys. Chem. C, 2010, 114, 11977–11983.
- 31 X. Lin, L. Wu, Y. Liu, A. L. Ong, S. D. Poynton, J. R. Varcoe and T. Xu, *J. Power Sources*, 2012, 217, 373–380.
- 32 C. H. Zhao, Y. Gong, Q. L. Liu, Q. G. Zhang and A. M. Zhu, *Int. J. Hydrogen Energy*, 2012, 37, 11383–11393.
- 33 A. I. Bochynska, T. G. Van Tienen, G. Hannink, P. Buma and D. W. Grijpma, *Acta Biomater.*, 2016, **32**, 1–9.
- 34 F. Scognamiglio, A. Travan, I. Rustighi, P. Tarchi, S. Palmisano, E. Marsich, M. Borgogna, I. Donati, N. de Manzini and S. Paoletti, *J. Biomed. Mater. Res., Part B*, 2016, **104**, 626–639.
- 35 S. Budun, E. İşgören, R. Erdem and M. Yüksek, *Appl. Surf. Sci.*, 2016, **380**, 294–300.
- 36 J. O. Vinhal, C. F. Lima and R. J. Cassella, *Ecotoxicol. Environ. Saf.*, 2016, **131**, 72–78.
- 37 N. G. Ireni, R. Narayan, P. Basak and K. V. S. N. Raju, *Polymer*, 2016, 97, 370–379.
- 38 R. Duarah, Y. P. Singh, B. B. Mandal and N. Karak, New J. Chem., 2016, 40, 5152–5163.
- 39 D. Filip, D. Macocinschi, E. Paslaru, C. G. Tuchilus and S. Vlad, *React. Funct. Polym.*, 2016, **102**, 70–81.

- 40 M. Sahraro, H. Yeganeh and M. Sorayya, *Mater. Sci. Eng.*, *C*, 2016, **59**, 1025–1037.
- 41 J. M. Spruell, M. Wolffs, F. A. Leibfarth, B. C. Stahl, J. Heo, L. A. Connal, J. Hu and C. J. Hawker, *J. Am. Chem. Soc.*, 2011, **133**, 16698–16706.
- 42 A. Saha, S. De, M. C. Stuparu and A. Khan, *J. Am. Chem. Soc.*, 2012, **134**, 17291–17297.
- 43 J. Guo, Z. Xie, R. T. Tran, D. Xie, D. Jin, X. Bai and J. Yang, Adv. Mater., 2014, 26, 1906–1911.
- 44 J. Hu, K. Peng, J. Guo, D. Shan, G. B. Kim, Q. Li, E. Gerhard, L. Zhu, W. Tu, W. Lv, M. A. Hickner and J. Yang, *ACS Appl. Mater. Interfaces*, 2016, 8, 17499–17510.
- 45 S. Karamdoust, B. Yu, C. V. Bonduelle, Y. Liu, G. Davidson, G. Stojcevic, J. Yang, W. M. Lau and E. R. Gillies, *J. Mater. Chem.*, 2012, 22, 4881.
- 46 J. J. Lin, W. C. Lin, S. D. Li, C. Y. Lin and S. H. Hsu, ACS Appl. Mater. Interfaces, 2013, 5, 433–443.
- 47 P. Wang, S. M. Zakeeruddin, M. Grätzel, W. Kantlehner, J. Mezger, E. V. Stoyanov and O. Scherr, *Appl. Phys. A*, 2004, 79, 73–77.
- 48 H. Xie, S. Zhang and H. Duan, *Tetrahedron Lett.*, 2004, 45, 2013–2015.
- 49 M. A. Gauthier, M. I. Gibson and H. A. Klok, Angew. Chem., Int. Ed. Engl., 2009, 48, 48–58.
- 50 J. Cheng, G. Yang, K. Zhang, G. He, J. Jia, H. Yu, F. Gai, L. Li, C. Hao and F. Zhang, *J. Membr. Sci.*, 2016, **501**, 100–108.
- 51 D. K. Chattopadhyay and D. C. Webster, *Prog. Polym. Sci.*, 2009, **34**, 1068–1133.
- 52 H. Sui, X. Ju, X. Liu, K. Cheng, Y. Luo and F. Zhong, Polym. Degrad. Stab., 2014, 101, 109–113.
- 53 K. Pielichowska, J. Bieda and P. Szatkowski, *Renewable Energy*, 2016, **91**, 456–465.
- 54 L. Shufen, J. Zhi, Y. Kaijun, Y. Shuqin and W. K. Chow, *Polym.-Plast. Technol. Eng.*, 2006, **45**, 95–108.
- 55 Z. S. Petrovic, Z. Zavarco, J. H. Flynn and W. J. Macknight, *J. Appl. Polym. Sci.*, 1994, **51**, 1087–1095.
- 56 K. G. Priyanka, A. K. Mishra, S. Kantheti, R. Narayan and K. V. S. N. Raju, *J. Appl. Polym. Sci.*, 2012, **126**, 2024–2034.
- 57 G. H. Zhao, Y. Liu, C. L. Fang, M. Zhang, C. Q. Zhou and Z. D. Chen, *Polym. Degrad. Stab.*, 2006, **91**, 703–711.
- 58 J. Y. Huang, H. Murata, R. R. Koepsel, A. J. Russell and K. Matyjaszewski, *Biomacromolecules*, 2007, 8, 1396–1399.
- 59 H. Murata, R. R. Koepsel, K. Matyjaszewski and A. J. Russell, *Biomaterials*, 2007, 28, 4870–4879.
- 60 M. J. Saif, J. Anwar and M. A. Munawar, *Langmuir*, 2009, 25, 377–379.
- 61 R. Kugler, O. Bouloussa and F. Rondelez, *Microbiology*, 2005, 151, 1341–1348.
- 62 A. M. Bieser and J. C. Tiller, *Macromol. Biosci.*, 2011, **11**, 526–534.
- 63 K. Lewis and A. M. Klibanov, *Trends Biotechnol.*, 2005, 23, 343–348.