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# Recent developments in 2D MXene-based materials for next generation room temperature NO<sub>2</sub> gas sensors

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MXenes with distinctive structures, good electrical conductivity and abundant functional groups have shown great potential in the fabrication of high performance gas sensors. Since the sensing mechanism of MXene-based gas sensors often involves a surface-dominant process, they can work at room temperature. In this regard, a significant amount of research has been carried out on MXene-based room temperature gas sensors and they can be viewed as one of the possible materials for NO<sub>2</sub> sensing applications in the future. In this review, we focus on the most recent research and improvements in pure MXenes and their nanocomposites for NO<sub>2</sub> gas sensing applications. First, we have explored the mechanisms involved in MXenes for NO<sub>2</sub> gas sensing. Following that, other ways to tune the MXene sensing performance are investigated, including nanocomposite formation with metal oxides, polymers, and other 2D materials. A comparative analysis of the RT NO<sub>2</sub> sensor performance based on MXenes and their hybrids is provided. We also discuss the major challenges of using MXene-related materials and the areas that can further advance in the future for the development of high-performance room temperature NO<sub>2</sub> gas sensors.

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## 1. Introduction

Human health has been severely harmed as a result of air pollution caused by urbanisation. Along with medical and technological advancements, the usage of synthetic fertilisers

for greater crop production has resulted in steady population growth while also increasing nitrogen dioxide (NO<sub>2</sub>) gas emissions into the atmosphere. NO<sub>2</sub> constitutes one of the most toxic gases, with a pungent odour, and it deteriorates human health when exposed to ppm levels for an extended period of time.<sup>1–3</sup> When compared to CO<sub>2</sub>, NO<sub>2</sub> is one of the main greenhouse gases that contribute to global warming and is responsible for stratospheric ozone depletion.<sup>4–6</sup> Hence, the

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design and fabrication of high performance NO<sub>2</sub> sensors operated at room temperature are very important to monitor the presence of low concentration gas molecules effectively. Accurate measurement of NO<sub>2</sub> gas under real environmental conditions at room temperature with high selectivity, and reversibility at low cost is a challenging task.<sup>7–9</sup>

In the past few decades, electrical signal-based NO<sub>2</sub> gas detection has been reported using sensors made of a variety of materials including metallic nanoparticles/metal-oxide,<sup>10–14</sup> polymers<sup>13,15–17</sup> and two-dimensional materials (2D) such as MoS<sub>2</sub>.<sup>18–21</sup> Recently, there has been a lot of interest in research on 2D materials, including MXenes, because of their properties like high surface area to volume ratio, layer-dependent tuneable mechanical, electrical, optical, and physicochemical properties arising from quantum confinement, and low dimensionality effects.<sup>22–24</sup> Among these 2D materials, graphene and transition metal dichalcogenide (TMDs)-based gas sensors are widely explored due to their excellent mechanical properties, high carrier mobility, and remarkable electrical and optical properties. Despite having an excellent sensor response and response time, the NO<sub>2</sub> sensors based on graphene suffered from long-recovery time whereas TMD-based sensors suffer from incomplete recovery due to its high adsorption.<sup>25</sup> This limitation motivated researchers to explore other 2D materials including MXenes. The interaction of gas molecules with sensing materials is an indelible feature of any gas-sensing process. Recently, MXene-based gas sensors have received a lot of attention due to their several advantages. Further, they have already shown applications in the fields of electrochemical energy storage devices, flexible electronic devices and so on due to their excellent thermal and

chemical stability. This family of 2D transition metal nitrides/carbides that possess intrinsic metallic conductivity demonstrate excellent gas sensing performance due to the properties ascribed above.<sup>26</sup> MXenes have the chemical formula of M<sub>n+1</sub>X<sub>n</sub>T<sub>x</sub> ( $n = 1–4$ ) where M represents the transition metal, X represents C or N, and T<sub>x</sub> corresponds to the functional groups –OH, –O, –F, etc.<sup>27,28</sup> MXene-based materials have attracted considerable interest for applications in gas sensing due to a variety of advantages such as high surface area, metallic conductivity, high mechanical flexibility, hydrophilicity, presence of abundant active sites, tuneable surface chemistry, and improved stability, among others.<sup>29–32</sup> Various approaches, such as doping, defect and vacancy engineering, heterostructure formation and modification with charge blocking layers and functional groups, and so on, are also used to improve the performance of the sensor in terms of its selectivity, limit of detection (LOD), response and recovery times, etc.

Among the various reported MXenes, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> is the most explored one for gas-sensing applications. Since the average thickness of the reported Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> layer (~2 nm) is much less than the depletion layer thickness, the sensing mechanism is expected to be a surface-dominated process. As a result, this kind of MXene-based gas sensor can operate at room temperature.<sup>33</sup> However, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> has drawbacks such as slow response kinetics and irreversibility that limit its use in RT gas sensor technology. Literature studies proved that modification of the constituents is not the only factor affecting the properties of MXene but functional group modification also plays an important role in determining its optical, mechanical and electrical properties. Theoretical studies proved that O-

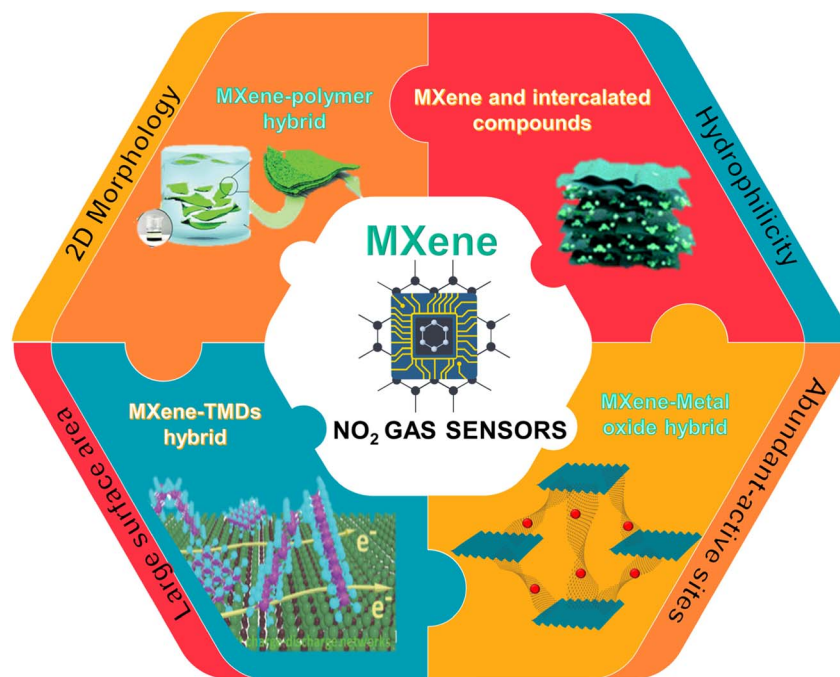


Fig. 1 A Scheme on the summary of the content of the review and with a part describing the advantages of MXenes for gas sensing, reprinted in part permission from *Adv. Funct. Mater.* 2020, 30, 0190302, Copyright (2023), Wiley Materials and *Adv. Funct. Mater.* 2020, 30, 2005305, Copyright (2023), Wiley Materials.



terminated MXenes are the best candidate for  $\text{NH}_3$  sensing due to their semiconductor electronic characteristics. Given the comparable atomic structures of the MXenes with different terminations, one wonders whether these new forms of MXenes could be used as  $\text{NO}_2$  gas sensors.<sup>34</sup> In this review, we focused on and discussed the recent literature on  $\text{NO}_2$  sensors operated at room temperature based on pristine and heterostructures of MXenes. We discussed the sensing mechanisms and different approaches to tune the sensing performance. Finally, the challenges and future perspectives of this research field for the development of high performance  $\text{NO}_2$  sensors are discussed. Fig. 1 summarizes the major contents of the review.

## 2. $\text{NO}_2$ gas sensing mechanisms of 2D materials and MXene-based sensors

In this section, we will discuss the mechanism involved in the MXene-based  $\text{NO}_2$  gas sensors.  $\text{NO}_2$  is generally a secondary product primarily generated from  $\text{NO}$  sources as given by eqn (1)



Because of the unpaired electron character of nitrogen in  $\text{NO}_2$ , it has an electron-accepting nature and acts as a strong oxidizing agent. Hence, the electrons from the sensing materials are taken by  $\text{NO}_2$  molecules.<sup>35</sup>

The sensing mechanisms of 2D material-based gas sensors are primarily explained using two well-established models. Specifically, (i) charge transfer mechanisms and (ii) the ion-adsorption model. In the case of charge transfer mechanisms,

electrons or holes act as the charge carriers depending on the type of materials (*i.e.*, p-type or n-type) being used as the active component of the sensor device (Fig. 2). Furthermore, the direction of the flow of charge transfer depends upon whether reducing or oxidising gas molecules are used as analytes. Additionally, the reactivity of the adsorbates and adsorbents, and their adsorption energy all affect how the analytes interact with the sensing materials.<sup>9,36,37</sup> The schematic representation of the charge transfer process used in 2D material-based gas sensors is shown in Fig. 2.<sup>38</sup>

MXenes have been investigated as an active material for gas sensing applications. As previously stated, the high conductivity, huge specific surface area, hydrophilicity and surface terminations of MXenes make them attractive for use in  $\text{NO}_2$  gas sensors.<sup>39</sup> The fundamental sensing mechanism of pure MXenes involved adsorption or desorption of gas analytes onto the sensing layer. The majority of the presented research elucidated the process involved in reducing gas sensing; however, they did not address the interaction of oxidising gases such as  $\text{NO}_2$ .<sup>32</sup> In 2017, it was discovered that  $\text{Ti}_3\text{C}_2\text{T}_x$  responds positively to reducing gases (methanol, ethanol, ammonia, and acetone). As a result, it was hypothesised that  $\text{Ti}_3\text{C}_2\text{T}_x$  was a p-type semiconductor and the gas sensing response was caused by the predominant charge carrier transfer by the interaction between the gas analyte and  $\text{Ti}_3\text{C}_2\text{T}_x$ .<sup>40</sup> In 2018 and 2019, it was found that MXene-based gas sensors always provide a positive response for all gases, regardless of their kind (oxidising or reducing).<sup>40–42</sup> Thus, p-type semiconductor technology is not the appropriate mechanism for MXene gas sensing. Therefore, two additional factors were presented to represent the positive



Fig. 2 A schematic representation of the charge transfer mechanism caused by gas adsorption on layered 2D materials. (a) The transport of electrons to and from adsorbent materials while gas molecules interact with the surface, depending on the distance, site of adsorption, gas type, molecule orientation and (b) charge transfer mechanism and density difference plots for  $\text{O}_2$ ,  $\text{H}_2\text{O}$ ,  $\text{NH}_3$ ,  $\text{NO}$ ,  $\text{NO}_2$ , and  $\text{CO}$  interacting with monolayer  $\text{MoS}_2$ , reproduced with permission from 2013 Yue *et al.*; license Springer.<sup>38</sup>



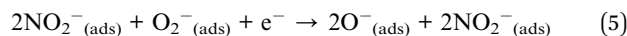
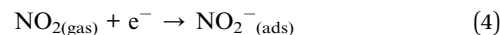
resistance change towards various gas analytes (i) MXene is a metallic compound rather than a semiconductor and this metallic sensing layer always hinders the charge-carrier transport.<sup>41</sup> This behaviour is completely distinct and is independent of the electron-donating/accepting properties of analytes and the dominant charge carrier type (*i.e.* p or n) of the sensing channel. In this context, (ii) interlayer spacing could be another cause for MXene's increased resistance to different gas analytes. Gas sensing, which occurs as a result of interlayer swelling after gas adsorption, impedes out-of-plane electron transport and increases electrical resistance.<sup>43</sup> The MXenes' metallic conductivity and interlayer spacing is a one-of-a-kind trait that occasionally makes the gas sensing process somewhat different and more exciting than the sensing mechanism of normal semiconducting materials. The surface functional groups also play a major role in the sensing mechanism as the hydrophilic group (–OH and =O) enhances the gaseous interaction with MXenes. Besides functional groups, terminal groups also affect the sensing mechanism. For example, O-terminated MXenes are the best candidates for NH<sub>3</sub> sensing due to their semiconductor electronic characteristics.<sup>32</sup> In other studies, done by Hu *et al.* they demonstrated that S-terminated MXene is the best candidate for NO<sub>x</sub> gas sensors.<sup>34</sup>

Besides the charge-transfer mechanism, Zhang *et al.* employed an ionosorption model to better understand the sensing mechanisms of V<sub>2</sub>CT<sub>x</sub> MXenes, which could also be applied to other MXenes and hybrids.<sup>44</sup> Here O<sub>2</sub><sup>–</sup>(<sub>ads</sub>) ions interact and contribute to sensing processes at room temperature or low temperatures. Because this review is about MXene-based room temperature sensors, O<sub>2</sub><sup>–</sup>(<sub>ads</sub>) oxygen ion species play an important role here, and other oxygen species contribute to gas sensors that work at higher temperatures.<sup>45–47</sup> O<sub>2</sub><sup>–</sup>(<sub>ads</sub>) oxygen ions are adsorbed over the surface of active sensor materials such as MXenes at ambient temperature *via* the following reactions:

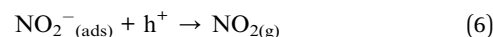


The width of the junction potential and the electron depletion layer at the grain boundaries increase when the gas sensors come in contact with an oxidising gas like NO<sub>2</sub>. This is because the gas molecules not only absorb electrons from the active

materials but also interact with the adsorbed oxygen ion species (O<sub>2</sub><sup>–</sup>(<sub>ads</sub>)). Following are several formulas for the chemical reaction caused by the interaction of NO<sub>2</sub> gas molecules at room temperature or at low temperatures (Fig. 3):



When air is introduced to gas sensors in the reverse process, NO<sub>2</sub><sup>–</sup> interacts with the holes and releases electrons once more, resulting in the formation of NO<sub>2</sub> gas molecules.



The above-given reactions are the essence of MXene-based RT NO<sub>2</sub> sensors. Fig. 3 describes the schematic illustration of the NO<sub>2</sub> sensing mechanisms based on the ionosorption model.<sup>48</sup>

Also, the synergistic and reverse enhancement effect in MXene composites directly influences the NO<sub>2</sub> sensing mechanism depending on the type and properties of the foreign materials. Generally, MXenes are reported as a channel layer and metal oxides are used as a supporting layer during composite formation. In this case, the basic mechanism is very similar to that of the conventional metal oxide semiconductor-based sensors in which effective absorption/desorption of the gas analyte on the surface of sensing materials shows an effective change in device resistance. The gas sensing mechanism of MXene-metal oxide composites is connected to the interfacial interactions and heterojunction development of both involved materials. The type of response (p- or n-type) following composite formation is entirely dependent on the composition of both materials as well as the material that plays the predominant role in carrier conduction.<sup>35</sup> Gasso *et al.* synthesized a WO<sub>3</sub>/Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> hybrid for NO<sub>2</sub> sensing where this composite shows electron transfer from MXene to WO<sub>3</sub> leading to the formation of heterojunctions. The oxygen adsorption on the surface leads to an electron depletion layer (EDL) and hole accumulation layer (HAL). When the sensor is exposed to NO<sub>2</sub> it captures electrons from the conduction band of WO<sub>3</sub> that lead to an increase in resistance.<sup>49</sup>

The gas sensing mechanism of oxidised MXenes is generally considered as a Schottky barrier (SB) modulation. The oxidizing tendency of Ti<sub>3</sub>C<sub>2</sub> flakes to form a heterojunction consisting of metallic Ti<sub>3</sub>C<sub>2</sub> and semiconducting TiO<sub>2</sub> is utilized to form the SB sites. Here Ti<sub>3</sub>C<sub>2</sub> MXene with intrinsic metallic conductivity offers the possibility of SB modulation within the sensing channel itself. Choi *et al.* demonstrated this *in situ* formation of multiple SBs in a single NO<sub>2</sub> gas sensing channel. In the presence of oxidising gas such as NO<sub>2</sub> SB can be shifted upwards which surpasses the transport of electrons and lead to a high gas-sensing response. The sensing mechanism for MXene-polymers also depends upon a number of factors including redox reactions between the gas analytes and hybrids, and charge carrier concentration changes happening in sensing

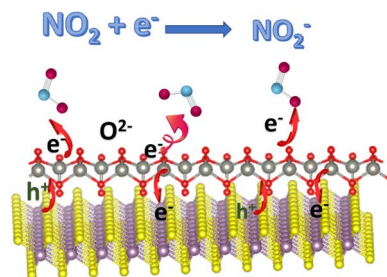


Fig. 3 NO<sub>2</sub> sensing mechanism based on an ionosorption model for 2D materials.



layers.<sup>50</sup> Zhao *et al.* reported the NO<sub>2</sub> sensing mechanism of a poly(L-glutamic acid) (γ-PGA)-MXene sensor which is different from the conventional metal oxide semiconductor-based sensors. Here, the gas molecules are adsorbed on the γ-PGA using a non-covalent bond. In the presence of NO<sub>2</sub>, the resistance of the sensor changes from a negative response to a positive response. Water molecules in the presence of air adsorbed onto the γ-PGA film and hydrolysed. At high concentrations of NO<sub>2</sub> excess molecules are adsorbed onto the surface *via* hydrogen bonding and electrostatic interaction and it may compete with H<sub>2</sub>O molecules for adsorption by the equation given below; this hinders ion conduction, thereby increasing the resistance. Here, the blocking behaviour of MXene was enhanced by γ-PGA and the acceleration rate was also increased.<sup>51</sup>



Aside from the synthesis method and etching conditions, the MXene surface termination has a considerable impact on NO<sub>2</sub> gas sensing. For example, it was recently found that NaOH alkalization can shift the response of the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> type to negative.<sup>52</sup> According to XPS measurements, alkalization increased the O/F ratio of the termination on this Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene from 2.6 to 7.6. As a result, it was claimed that a high O/F termination ratio can change the response type from positive to negative. However, the underlying reasons behind this conversion remain unknown, and it is not clear whether this conversion occurs in other MXenes.<sup>52</sup> According to Zhang *et al.*'s studies, this alkalinization treatment with DMSO improves the NO<sub>2</sub> gas sensing performance of V<sub>2</sub>CT<sub>x</sub> MXene, indicating that this conversion also occurs with other MXenes.<sup>44</sup> They demonstrated that the adsorbed H<sub>2</sub>O and O results in p-type sensing behaviour in V<sub>2</sub>CT<sub>x</sub>. With the exposure of V<sub>2</sub>CT<sub>x</sub> MXene-based sensors to humid air, the oxygen molecules absorbed on the surface of V<sub>2</sub>CT<sub>x</sub> ionized to O<sub>2</sub><sup>-</sup> with the consumption of electrons. As a result, the concentration of the V<sub>2</sub>CT<sub>x</sub> MXene major charge carrier increases, increasing conductivity and decreasing resistance. When NO<sub>2</sub> molecules are exposed to the V<sub>2</sub>CT<sub>x</sub> MXene-based sensors, they are adsorbed on the active sites by surface terminations such as -OH and -O. Electrons can be transferred from the V<sub>2</sub>CT<sub>x</sub> to NO<sub>2</sub> gas molecules. This charge transfer results in increasing the hole concentration of MXene, which further lowers the resistance and the conductivity of V<sub>2</sub>CT<sub>x</sub> MXene-based sensors increase. The studies carried out by Choi *et al.* also prove that the response of Mo<sub>2</sub>CT<sub>x</sub> MXene using tetramethylammonium hydroxide (TMAOH) as the intercalant can change the response type which can be attributed to the high density of MXene surface functional groups and its intrinsic metallic conductivity.<sup>53</sup> The DFT studies demonstrated that TMA intercalated MXenes show high adsorption energy towards NO<sub>2</sub>

### 3. Recent advancements in NO<sub>2</sub> gas sensors using MXene-based materials

MXenes are an excellent choice for gas-sensing applications due to their abundant active sites, large aspect ratio, availability of

abundant functional groups, hydrophilicity, metallic conductivity and tunable surface chemistry. However, MXenes show disadvantages such as irreversibility and low response kinetics, which limit their gas sensing applications.<sup>51</sup> The development of MXene heterostructures to modify the physiochemical properties of MXenes is a possible route for the development of high-performance low-temperature sensors. Surface modification, functionalization with noble metals, additive doping, inorganic heterojunction sensitization, light activation, *etc.* are some of the other approaches which can further tune the properties of MXenes and make them better candidates for RT NO<sub>2</sub> gas sensing. MXenes have a massive gas adsorption area, variable layer numbers, and an interlayer swelling effect, making them an amazing gas-sensitive material with a good signal-to-noise ratio.<sup>44,54-56</sup> Table 1 summarizes the reported NO<sub>2</sub> sensing performance of different MXenes and their hybrid-based sensors.

#### 3.1. Pristine and intercalated MXenes

The theoretical studies conducted by Yu *et al.* first revealed the possibilities of MXenes for gas sensing applications. Their DFT analyses demonstrate that other gas molecules, notably NO<sub>2</sub>, display distinct adsorption behaviours compared to NH<sub>3</sub>. Lower adsorption energies of NO<sub>2</sub> gas molecules on the Ti<sub>2</sub>CO<sub>2</sub> monolayer reflect weaker interactions between MXene and NO<sub>2</sub>, implying that this Ti<sub>2</sub>CO<sub>2</sub> MXene is unsuitable for NO<sub>2</sub> gas sensing applications.<sup>57</sup> Furthermore, Jian *et al.* revealed that the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene-based gas sensor is not suited for strong and moderate oxidising gases such as NO<sub>2</sub> due to its high proclivity to oxidise to TiO<sub>x</sub>.<sup>33</sup>

The other pristine MXenes which are reported to show promising NO<sub>2</sub> sensing performance are Mo<sub>2</sub>CT<sub>x</sub>, Nb<sub>2</sub>CT<sub>x</sub>, and V<sub>2</sub>CT<sub>x</sub>.<sup>44,53,56,58,59</sup> For example, Molybdenum carbide-based NO<sub>2</sub> sensors demonstrated a high signal-to-noise ratio with the ability to detect NO<sub>2</sub> concentration even at the ppb level and high ambient stability due to their high electrical conductivity, rich density of states near the Fermi level, superior catalytic properties, good resistance to corrosion and low chemical reactivity.<sup>59</sup> According to reports, Mo<sub>2</sub>CT<sub>x</sub> MXene exhibits a three-phase transition to gas sensing behaviour. Here, the film thickness and the presence of organic intercalants play a crucial role in tuning the performance.<sup>53</sup> It is demonstrated that the thin film sensor with 5 nm thick Mo<sub>2</sub>CT<sub>x</sub> intercalated by tetramethylammonium hydroxide (TMAOH) shows a p-type gas sensing response whereas it displayed an n-type response without intercalants and films with thickness above 700 nm exhibit conductor type response. Because of the stronger gas molecule binding in the presence of the intercalants, the intercalation technique allows the gas molecules to penetrate, resulting in increased gas sensing performance with sensitivity 30 times greater than that of the deintercalated films. The effect of the thickness of the film is also studied and it demonstrates that the thin film of Mo<sub>2</sub>CT<sub>x</sub> shows a better response towards NO<sub>2</sub>. This is because as the thickness of the film reduces, gas molecules can more easily penetrate into the interlayer space, adsorbing on the surface of MXene flakes faster and resulting in a quick response and short recovery time.<sup>53</sup>



Table 1 Comparison table showing the performance of MXene-based NO<sub>2</sub> gas sensors at room temperature

Materials	Dynamic range	Sensor performance, $S = R_a/R_g$ , $t_{res}$ = response time, $t_{rec}$ = recovering time, LOD = limit of definition	Sensing environment (temp., dry air, RT, etc.)	Reference
Mo <sub>2</sub> CT <sub>x</sub>	0.125–5 ppm	$S = 18.2$	RT, N <sub>2</sub>	53
Nb <sub>2</sub> CT <sub>x</sub>	5–25 ppm	LOD = 67 ppb $t_{res} = 39$ s $t_{rec} = 66$ s	RT, air	56
Nb <sub>2</sub> CT <sub>x</sub> -CTAB	5–25 ppm	$S = 16.01$ for 25 ppm LOD = 21 ppb $t_{res} = 39$ s $t_{rec} = 78.9$ s	RT, air	56
V <sub>2</sub> CT <sub>x</sub>	5–50 ppm	$S = 50.23$ for 25 ppm $t_{res} = 76$ s $t_{rec} = 20$ s	25 °C/RT, air	60
SnO <sub>2</sub> -MXene	1–960 ppb	$S = 70$ for 50 ppm $S = 23\%$ for 30 ppb $t_{res} = 146$ s $t_{rec} = 102$ s	RT, air	68
Microwave irradiated SnO <sub>2</sub> -MXene	0.1–10 ppm	$S = 24.8$ for 10 ppm LOD = 0.0153 ppm	150 °C, air	67
Facet controlled SnO <sub>2</sub> /Ti <sub>3</sub> C <sub>2</sub>	0.05–10 ppm	Response = $(\Delta R/R_a) = 0.02$ –1.57	25 °C/RT, air	66
Self-powered SnO <sub>2</sub> fibre/sodium L-ascorbate-treated MXene	—	$t_{res} = 265$ ms $t_{rec} = 75.5$ ms	RT, air	69
Ti <sub>3</sub> C <sub>2</sub> /TiO <sub>2</sub>	0.125–5 ppm	LOD = 0.03 ppb NO <sub>2</sub> $S = 16\%$ for 5 ppm LOD = 125 ppb	RT, air	53
ZnO-Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> UV excited	5–200 ppb	$S = 81\%$ for 50 ppm LOD = 0.2 ppb $t_{res} = 17$ s $t_{rec} = 24$ s	RT, air	70
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> -ZnO sphere	5–100 ppm	$S = 41.93\%$ for 100 ppm $t_{res} = 53$ s $t_{rec} = 103$ s	RT, air	71
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> -ZnO nanosheets	—	$S = 367.63\%$ to 20 ppm $t_{res} = 22$ s $t_{rec} = 10$ s	RT, air	73
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> -CuO	1–50 ppm	$S = 56.9\%$ for 50 ppm $t_{res} = 16.6$ s $t_{rec} = 31.3$ s	RT, air	74
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> /TiO <sub>2</sub> /rGO	10–500 ppb	19.85% for 5 ppm	RT, air	26
WO <sub>3</sub> /Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	15–500 ppb	$t_{res} = 182$ s $t_{rec} = 75$ s	RT, air	49
(Self-powered) Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> /WO <sub>3</sub>	0.5–50 ppm	$S = 510\%$ for 50 ppm	RT, air	75
Ti <sub>3</sub> C <sub>2</sub> -WO <sub>3</sub>	30–1000 ppb	$S = 19\%$ for 30 ppb $t_{res} = 25$ s $t_{rec} = 40$ s	RT, air	76
BiOCl-MXene	100 ppm	$S = 34.58$ $t_{res} = 3.15$ s $t_{rec} = 31.05$ s	RT	77
MXene-PEODT/PSS	1 ppm	—	65 °C	91
CO <sub>3</sub> O <sub>4</sub> @PEI/Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	0.3–100 ppm	$S = 27.9\%$ @ 100	RT, air	78
$\alpha$ -MOC <sub>1–x</sub>	0.125–5 ppm	$S = 15$ LOD = ppb–ppt level	RT, air	85
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> /WS <sub>2</sub>	0.1–20 ppm	$S = 15.2\%$ @ 1 ppm LOD = 11 ppb $t_{rec} = 60$ s	RT, air	86
2H MoS <sub>2</sub> /TiC <sub>3</sub> T <sub>2</sub> T <sub>x</sub>	30–70 ppm	65.6% @ 100 ppm	RT, air	18
MoS <sub>2</sub> /Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	1–50 ppm	40% @ 20 ppm $t_{res} = 525$ s $t_{rec} = 155$ s	RT, air	19
Mo <sub>2</sub> TiC <sub>2</sub> T <sub>x</sub> /MoS <sub>2</sub>	0.2–50 ppm	$S = 415.8\%$ @ 50 ppm LOD = 2.5 ppb $t_{res} = 33.5$ s $t_{rec} = 140.1$ s	RT, air	21



Table 1 (Contd.)

Materials	Dynamic range	Sensor performance, $S = R_a/R_g$ , $t_{res}$ = response time, $t_{rec}$ = recovering time, LOD = limit of definition	Sensing environment RT = room temperature (temp., dry air, RT, etc.)	Reference
Black phosphorous QDS/Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub>	50 ppb–10 ppm	LOD = 13 ppb $t_{res}$ = 72 s $t_{rec}$ = 85 s	RT, air	90
Nb <sub>2</sub> CT <sub>x</sub> -APTES	5–25 ppm	$S = 31.52\%$ @ 25 ppm LOD = 15 ppb LOQ = 52 ppb $t_{res}$ = 36 s $t_{rec}$ = 96.8 s	RT, air	65
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> /γ-PGA	2–50 ppm	$S = 1127.3\%$ $t_{res}$ = 43.4 s $t_{rec}$ = 3 s	RT, air	96
Ti <sub>3</sub> C <sub>2</sub> /TiO <sub>2</sub>	1–100 ppm	$S = 19.76$ for 100 ppm	RT, air	98
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> @TiO <sub>2</sub> @MoS <sub>2</sub>	0.02–50 ppm	$S = 55.16$ ( $R_a/R_g$ ) @ 50 ppm $t_{res}$ = 1.8 s $t_{rec}$ = 50 s LOD = 23 ppb	RT, air	98
MXene-derived TiO <sub>2</sub> -rGO	0.05–20 ppm	$S = 400\%$ for 20 ppm LOD = 50 ppb $t_{res}$ = 130 s $t_{rec}$ = 230 s	RT, air	99
MXene-derived TiO <sub>2</sub> -SnS <sub>2</sub>	—	115 for 1000 ppm NO <sub>2</sub>	RT, air	100

Rathi *et al.* employed Nb<sub>2</sub>CT<sub>x</sub> to demonstrate that the strong metallic conductivity of MXenes is beneficial for noise reduction and that the abundance of functional groups is favourable for achieving higher sensitivity.<sup>56</sup> It has been found that the sensor response of Nb<sub>2</sub>CT<sub>x</sub> MXenes treated with cetyltrimethylammonium bromide (CTAB) increases thrice when compared to pure MXenes. The activation of the exposed surface, as well as the interlayer swelling with the hydroxyl groups, play an important role in the adsorption of target molecules, resulting in increased sensing characteristics for delaminated MXene by CTAB.

V<sub>2</sub>CT<sub>x</sub> MXenes intercalated by Na<sup>+</sup> ions also show 80 times higher response than the as-prepared samples for NO<sub>2</sub> gas (5–50 ppm).<sup>60</sup> Intercalation by Na atoms swelled the layers of the MXenes which allowed the analyte gases to enter inside for better interaction. Further, the existence of the surface functional groups (–OH) allowed the adsorption of water molecules on MXenes leading to more reactions with NO<sub>2</sub> to form NO (3NO<sub>2</sub> + H<sub>2</sub>O → 2HNO<sub>3</sub> + NO). Hence, the swelling effect and surface adsorption promoted by the functional groups present in MXenes contributed significantly to achieve enhanced NO<sub>2</sub> sensing performance.<sup>60</sup>

**3.1.1. MXene modification.** Studies revealed that MXenes can be stored for a long time at low temperature, whereas their stability is very low in aqueous solution at RT or high temperature. Thus, to investigate the electrical characteristics and environmental stability of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>, Lipatov and colleagues fabricated field-effect transistors (FETs) with single-layered Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> flakes as the conductive channels. The results showed that single-layered Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> flakes had more field-effect electron mobility and resistivity than bulk Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>. According to the

environmental stability data, these FETs based on Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> remain stable and highly conductive even after 70 hours of exposure to humid air.<sup>61</sup> Even though MXenes have great sensitivity and detection limits, their stability in oxidative environments, which is rarely discussed in the literature, is a key limitation in the fabrication of RT-based NO<sub>2</sub> sensors. MXenes will rapidly degrade over time in the presence of air or in the presence of water and their hydrophilic properties make them more vulnerable to oxidation.<sup>35</sup> For example, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> undergoes oxidation while annealing at high temperatures and also under plasma treatment and results in the formation of TiO<sub>2</sub> which possesses higher oxygen adsorption capacity and electronic transmission.<sup>62,63</sup> The functional groups also strongly affect the electronic properties as well as the work function of MXenes. With the help of inverse photoelectron spectroscopy, May *et al.* studied the work function variation of MXenes with the functional groups. Their research demonstrated that heating MXenes in a vacuum atmosphere raises their work function from 3.9 to 4.8 due to water adsorption, –OH species, and carbon-dominated contaminations.<sup>35,63</sup> These studies prove that MXene stability is very important during the fabrication of gas sensors.

Compared to other MXenes Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> is not much preferred for oxidising gases such as NO<sub>2</sub>, because it gets easily oxidised to TiO<sub>2</sub>. The studies done by Jian *et al.* showed that Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>-based gas sensors give stable response–recovery curves for both reducing and oxidizing gases where this sensor shows a major baseline resistance drift in the case of NO<sub>2</sub> gas sensing. This irreversible performance towards NO<sub>2</sub> gas could be attributed to Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> oxidation in an oxidising environment. Hence, compared to other MXenes Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> is not much preferred for



oxidising gases such as  $\text{NO}_2$ , because it gets easily oxidised to  $\text{TiO}_2$ .<sup>33</sup> In this scenario, Chae *et al.* stored  $\text{Ti}_3\text{C}_2\text{T}_x$  in an aqueous solution at different temperatures and aged for different time intervals. Their studies concluded that MXene which is stored at  $-80^\circ\text{C}$  for 5 weeks and as-synthesized samples show the same response towards  $\text{NO}_2$  at 5 ppm demonstrating that the

oxidation stability can be controlled by maintaining the storage conditions.<sup>64</sup> To address these issues of oxidation, several techniques including modification with a hydrophobic fluorosilane layer, and polymers were done. This surface modification helps to decrease MXene oxidation and also allows the simultaneous introduction of additional reactive groups. For

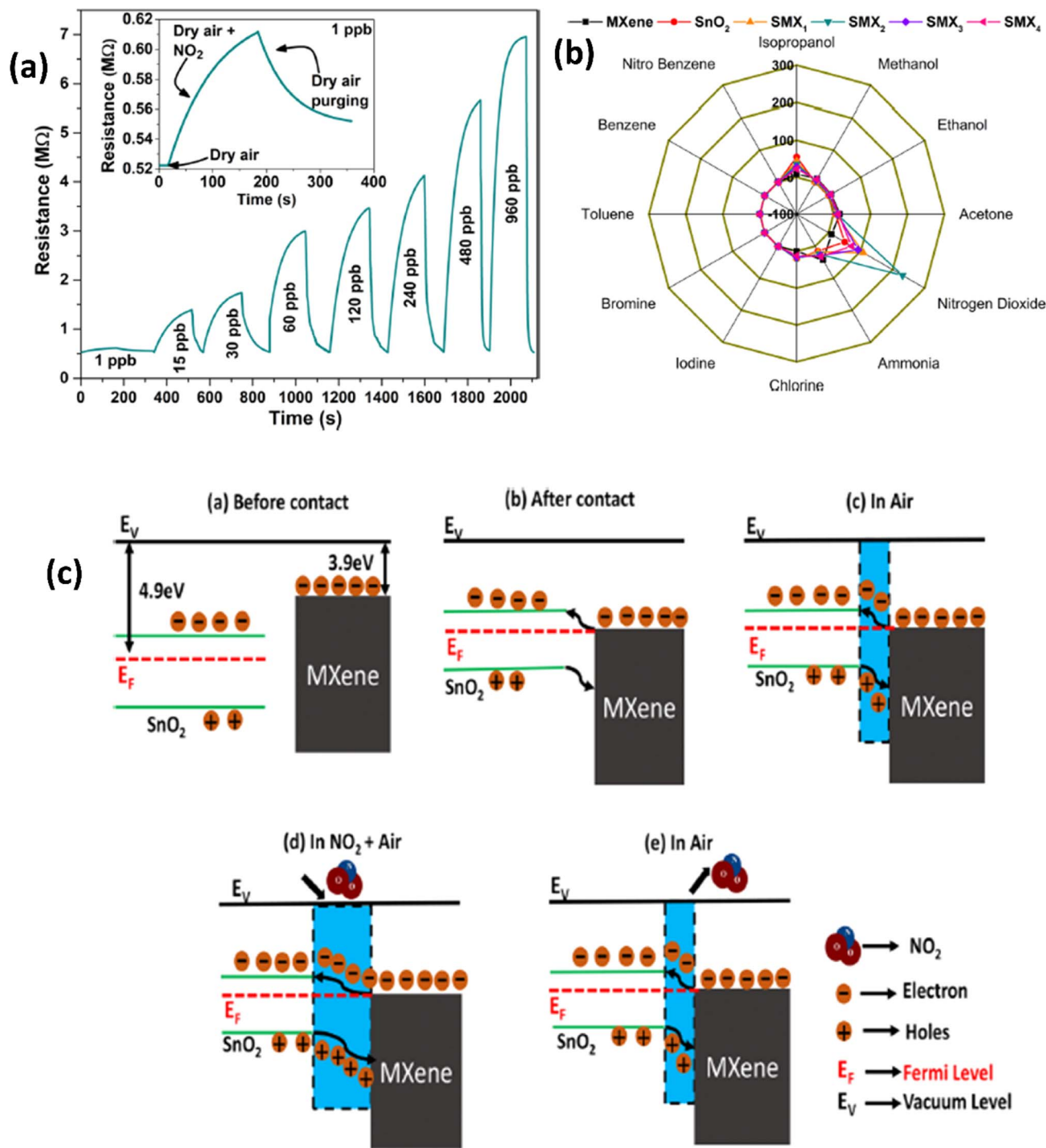


Fig. 4 (a) Resistance–time curves for the  $\text{SnO}_2/\text{MXene}$  sensor with exposure to different concentrations of  $\text{NO}_2$ . (b) Polar plots for various gases at 30 ppb concentrations for pristine  $\text{SnO}_2$ , MXenes and their heterostructures at room temperature. (c) Schematic representation of the  $\text{NO}_2$  sensing mechanism of the  $\text{SnO}_2/\text{MXene}$  heterostructure, reprinted in part permission from ref. 60 Copyright (2023) Elsevier.





example, Naveen Kumar *et al.* demonstrated that the introduction of amine groups over the  $\text{Nb}_2\text{CT}_x$  MXene helps in the detection of acidic gases such as  $\text{NO}_2$  by acting as an electron acceptor.<sup>65</sup> Rathi *et al.* also modified  $\text{Nb}_2\text{CT}_x$  MXene with CTAB and here the stability enhancement was ascribed to the (i) non-covalent bonding between CTAB and  $\text{Nb}_2\text{CT}_x$  MXene (ii) the long hydrophobic chains in CTAB hinders the interaction of humidity and oxygen with  $\text{Nb}_2\text{CT}_x$  MXene. CTAB functionalization can aid in the bulk manufacture of stable MXenes under ambient conditions for future device manufacturing in  $\text{NO}_2$  gas sensors.<sup>56</sup>

### 3.2. MXene-based hybrid materials

**3.2.1. MXene-metal oxide hybrids.** To avoid restacking of MXene sheets and to retain the high surface area, hybrid materials with metal oxides have been explored to achieve improved  $\text{NO}_2$  sensing properties.<sup>66–68</sup> Due to the presence of abundant surface adsorption species and high gas sensing capabilities, metal oxides such as  $\text{CuO}$ ,  $\text{TiO}_2$ ,  $\text{WO}_3$ ,  $\text{BiOCl}$ ,  $\text{Co}_3\text{O}_4$ , *etc.* have been used to fabricate hybrid materials with

MXenes, and they displayed promising  $\text{NO}_2$  sensing properties. Gasso *et al.* revealed that by regulating the  $\text{SnO}_2$  loading in the MXene- $\text{SnO}_2$  heterostructures, high sensor performance in terms of selectivity, sensitivity, reproducibility, and repeatability can be achieved under humid conditions.<sup>68</sup> Sensors based on 20 wt%  $\text{SnO}_2$  in MXene- $\text{SnO}_2$  heterostructures had a nearly 5 times greater response (231%) to 30 ppb  $\text{NO}_2$  at ambient temperature, with a quicker recovery time (146 s) and response time (102 s) than pure  $\text{SnO}_2$  (Fig. 4). In another report, it is shown that the microwave-irradiated  $\text{SnO}_2$  (2 wt%)- $\text{Ti}_3\text{C}_2$  nanocomposites exhibited a superior response of 24.8 for 10 ppm  $\text{NO}_2$  as compared to the pristine MXene,  $\text{SnO}_2$  and un-irradiated  $\text{SnO}_2$ -MXene nanocomposite-based gas sensors.<sup>67</sup> Liu *et al.* reported that by regulating the co-exposed (110) and (221)  $\text{SnO}_2$  facets in the MXene- $\text{SnO}_2$  nanocomposites, excellent sensing properties in terms of linear response ( $R^2 = 0.99729$ ), good selectivity and recycling performance can be achieved.<sup>66</sup>  $\text{NO}_2$  sensing mechanisms of  $\text{SnO}_2$ -MXene hybrids involve a charge transfer process where the adsorbed oxygen ion species played an important role as per eqn (2) and (3) (Fig. 4c). These adsorbed oxygen ions trap electrons from the  $\text{SnO}_2$  conduction

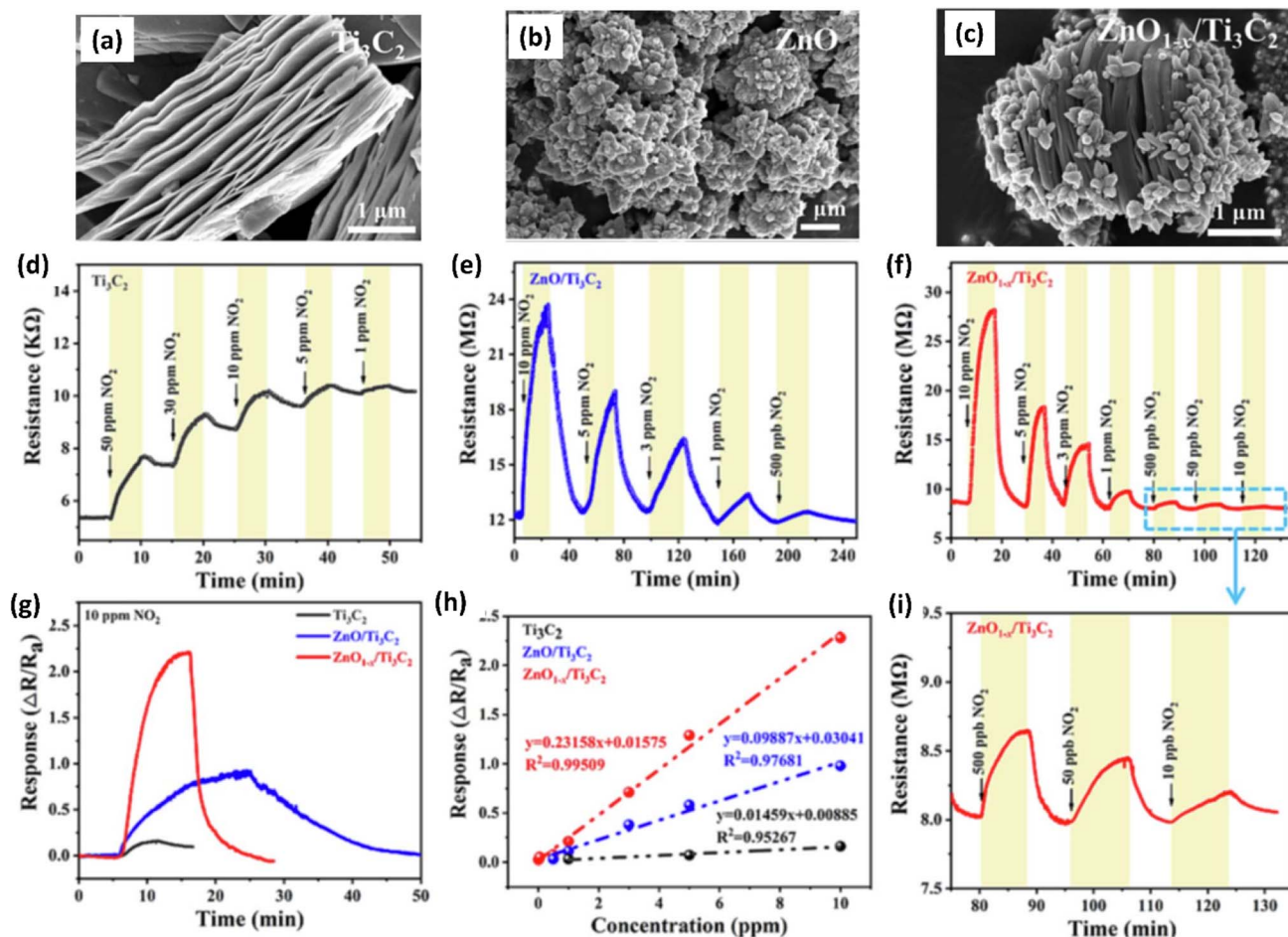


Fig. 5 SEM images showing morphology of (a) pristine  $\text{Ti}_3\text{C}_2$  (MXene), (b) ZnO and (c)  $\text{ZnO}/\text{Ti}_3\text{C}_2$  (MXene), (d–f) Response/recovery curves of  $\text{ZnO}/\text{Ti}_3\text{C}_2$  (MXene) (g) response curves and (h) linear fitting curves of  $\text{Ti}_3\text{C}_2$  (MXene) and  $\text{ZnO}/\text{Ti}_3\text{C}_2$  (MXene) heterostructures at room temperature to gas  $\text{NO}_2$ . (i) Response/recovery curves of  $\text{ZnO}/\text{Ti}_3\text{C}_2$  (MXene) heterostructures at different concentrations of  $\text{NO}_2$  (10, 50 and 500 ppb), reprinted with permission from ref. 63 from Copyright (2023) Elsevier.



band, forming the electron depletion layer (EDL). Due to their mismatched work function ( $\text{SnO}_2 = 4.9 \text{ eV}$  and MXene =  $3.9 \text{ eV}$ ) and Fermi level locations, metallic MXene and semiconducting  $\text{SnO}_2$  heterojunctions generate Schottky barriers. To reach equilibrium, electrons from MXene move to  $\text{SnO}_2$  until equilibrium is reached, resulting in band bending. When exposed to air, the adsorbed oxygen species ( $\text{O}_2^-$ ) form the EDL and hole accumulation layer (HAL) at the heterojunction interface, and these layers interact with the  $\text{NO}_2$  gas as shown in eqn (4) and (5). Finally,  $\text{NO}_2$  molecules take electrons from the heterostructure's surface, increasing the EDL width and trapping a high number of holes in the HAL, thus lowering the total conductivity. Gasso *et al.* recently developed a self-powered humidity-tolerant gas sensor made up of MXene treated with sodium L-ascorbate and  $\text{SnO}_2$  nanofibers. This  $\text{SnO}_2/\text{MXene}$  nanocomposite shows excellent response towards  $\text{NO}_2$  when

compared to pristine MXene and  $\text{SnO}_2$  with a limit of detection around  $0.3 \text{ ppb}$ .<sup>69</sup>

Due to their high surface area, sizable active sites, and modification of the carrier density, MXene-ZnO composites are another type of MXene-metal oxide composite that have been reported to exhibit promising  $\text{NO}_2$  sensing capabilities.<sup>14,70-72</sup>  $\text{ZnO}/\text{Ti}_3\text{C}_2\text{T}_x$  MXene nanocomposites displayed an improved response of 3.4 for  $\text{NO}_2$  (8 ppm) with a recovery time of 254 s and response times of 191 s respectively.<sup>14</sup> The crumpled spheres of the heterostructures demonstrated improved  $\text{NO}_2$  sensing performance due to their increased surface area, an abundance of edges and flaws generated by folding, and the development of the MXene/ $\text{ZnO}$  p-n junction.<sup>71</sup> This heterostructure showed enhanced response from 27.3% to 41.9% for 100 ppm  $\text{NO}_2$ , as well as a significant increase in the recovery rate from 30% to 100%.  $\text{ZnO}_{1-x}/\text{Ti}_3\text{C}_2\text{T}_x$  MXene composites

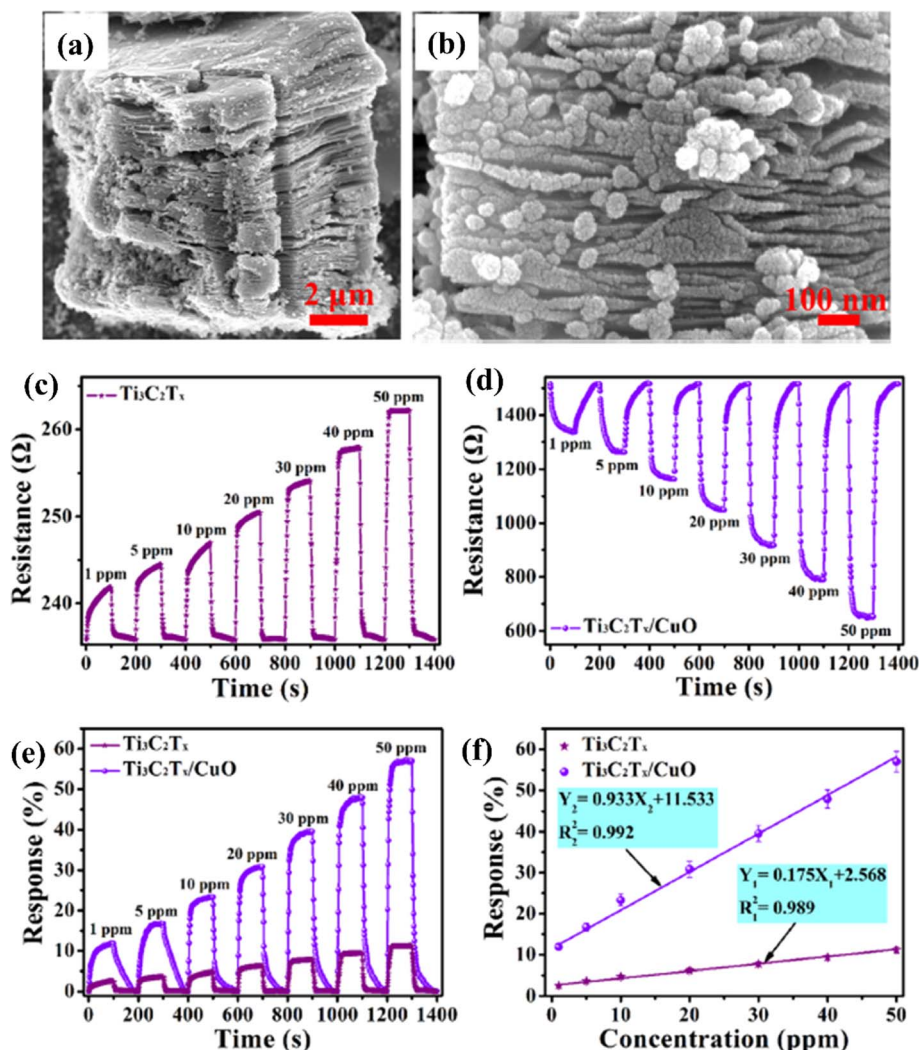


Fig. 6  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene)/CuO heterostructure for  $\text{NO}_2$  gas-sensing applications: (a and b) low- and high-resolution FESEM images of the  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene)/CuO heterostructure, (c and d) dynamic resistance curves of  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) and  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene)/CuO heterostructure-based sensors when exposed to  $\text{NO}_2$  with different concentrations varying from 1 to 50 ppm at  $23 \text{ }^\circ\text{C}$ , (e) normalized response curves of  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) and  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene)/CuO heterostructure-based sensors, (f) normalised response curves of  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) and  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene)/CuO heterostructure-based gas sensors at  $23 \text{ }^\circ\text{C}$  as a function of  $\text{NO}_2$  gas concentration, reprinted with permission from ref. 67 from Copyright (2023) Elsevier.



with abundant oxygen vacancies showed 2.3 and 14.2-times higher response as compared to the ZnO/Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> and pristine Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> respectively with good linear response ( $R^2 = 0.99509$ ), long-term stability and reproducibility<sup>14</sup> (Fig. 5). The sensing mechanisms for the ZnO-MXene heterostructure are predicted to follow a similar mechanism to that of the SnO<sub>2</sub>-MXene, in which energy bands bend and a Schottky barrier forms at the interface of the two materials.

Room temperature recovery is a big challenge for MXene-based NO<sub>2</sub> gas sensors. Thus, pristine MXene-based gas sensors experience incomplete recovery at room temperature which demands sensor operation at a higher temperature. However, thermal treatment for obtaining full recovery is not suitable. Recently, light-assisted recovery of gas sensors has opened up a new prospective avenue for developing RT gas sensors. Light illumination not only aids in sensor recovery but also improves 3S performance (recovery time, low response and

sensor response).<sup>25</sup> Based on these, Wang *et al.* showed that under UV illumination, the NO<sub>2</sub> sensing efficiency of MXene/ZnO nanorods can be considerably improved due to photo-generated electrons in ZnO reducing the depletion layers and enhancing the conduction path. The sensing response ranged from 21% to 346% for 5–200 ppb NO<sub>2</sub> at room temperature, with response and recovery times of 17 s and 24 s for 50 ppb NO<sub>2</sub> respectively.<sup>70</sup> In the regime of portable, flexible and wearable gas sensors self-powered sensors have garnered a lot of attention. Similarly, Fan *et al.* demonstrated that MXene/ZnO nanaosheet-based sensors show enhanced performance in the presence of UV illumination. They found out that the main adsorption site for NO<sub>2</sub> was present on the surface of ZnO nanosheets, while the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene plays a major role as a conductive path which helps to accelerate the charge carrier transformation.<sup>73</sup>

Because of the p-type conductivity ( $\sim 1.5$  eV) and narrow band gap, CuO is employed to fabricate p-n heterojunctions of

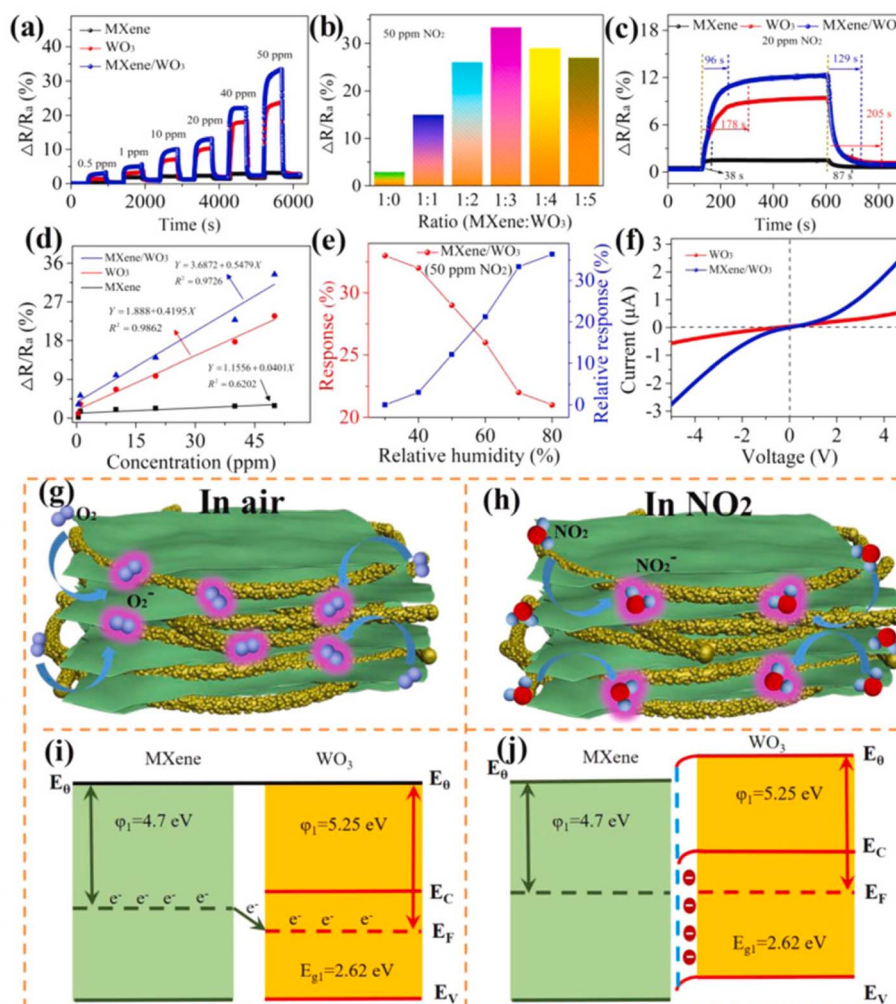


Fig. 7 MXene/WO<sub>3</sub> heterostructure for NO<sub>2</sub> gas sensing: (a) the dynamic response variation of the pristine MXene, WO<sub>3</sub> and MXene/WO<sub>3</sub> heterostructure-based sensors at different NO<sub>2</sub> concentrations. (b) The sensor response of composite materials with different mass ratios. (c) The response and recovery time of the MXene/WO<sub>3</sub> heterostructure-based sensor. (d) Response-concentration fitting curves of the three developed sensors. (e) The humidity effect on the MXene/WO<sub>3</sub> sensor. (f) The  $I$ - $V$  curves of WO<sub>3</sub> and the MXene/WO<sub>3</sub> sensor. Schematic of the gas-sensing mechanism and energy band structure of the MXene/WO<sub>3</sub> heterostructure (g and i) in the presence of air and (h and j) in the presence of NO<sub>2</sub> gas, reprinted with permission from ref. 69 from Copyright (2023) Elsevier.



CuO/MXene hybrids for NO<sub>2</sub> sensing applications.<sup>74</sup> (Fig. 6). For 50 ppm NO<sub>2</sub>, the response of mesoporous MXene-CuO nanocomposites was 5 times higher (56.99%) than pure MXene (11.7%), with ultra-fast response (16.6 s) and recovery time (31.3 s) to 20 ppm NO<sub>2</sub> and notable reversibility (over 40 days).

Since, theoretical calculations predicted that both Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene and WO<sub>3</sub> have significant NO<sub>2</sub> reactivity with adsorption energies ( $E_{\text{ads}}$ ) of -1.12 and 0.54 eV respectively, researchers have investigated NO<sub>2</sub> sensing performance of WO<sub>3</sub>/Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene.<sup>49,75,76</sup> Further, MXene sheets treated with sodium L-ascorbate in WO<sub>3</sub>/Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene hybrids are reported to show enhanced reversibility and stability under varying humid conditions (0–99% RH).<sup>49</sup> The sensing mechanisms of MXene-WO<sub>3</sub> hybrids are reported to be similar to the proposed mechanisms for other metal oxide-MXene composites. For the MXene-WO<sub>3</sub> heterostructures, the MXene platform for WO<sub>3</sub> nanorods significantly restricted the aggregation of WO<sub>3</sub> and helped to achieve increased interfacial contacts, enhanced

surface area for the adsorption of gas and quicker charge transit. Wang *et al.* reported a high performing self-powered Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene/WO<sub>3</sub> sensor powered by triboelectric nano-generators (TENGs) with a response of 510% for 50 ppm NO<sub>2</sub>, which was 15 times greater than that of a resistive MXene/WO<sub>3</sub> sensor<sup>75</sup> (Fig. 7). Heterostructures based on the highly conductive Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene and p-type semiconductor BiOCl offered electronic transmission channels with excellent response and quick response/recovery periods, as well as a lower detection limit (50 ppb) for NO<sub>2</sub> gas.<sup>77</sup> Sun *et al.* constructed a NO<sub>2</sub> sensor based on Co<sub>3</sub>O<sub>4</sub> nanocrystals decorated on the surface of poly-ethylenimine (PEI) sheet functionalized MXene.<sup>78</sup> MXene provided the electron transport channels whereas high surface area and active sites of the p-type Co<sub>3</sub>O<sub>4</sub> nanocrystals contributed to achieving high NO<sub>2</sub> sensing performance with good response and recovery times of 27.9 s and 2 s for NO<sub>x</sub> gas at RT and at 26% RH, excellent selectivity, reproducibility and an LOD of 30 ppb. Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/TiO<sub>2</sub>/rGO heterostructure-based sensors have

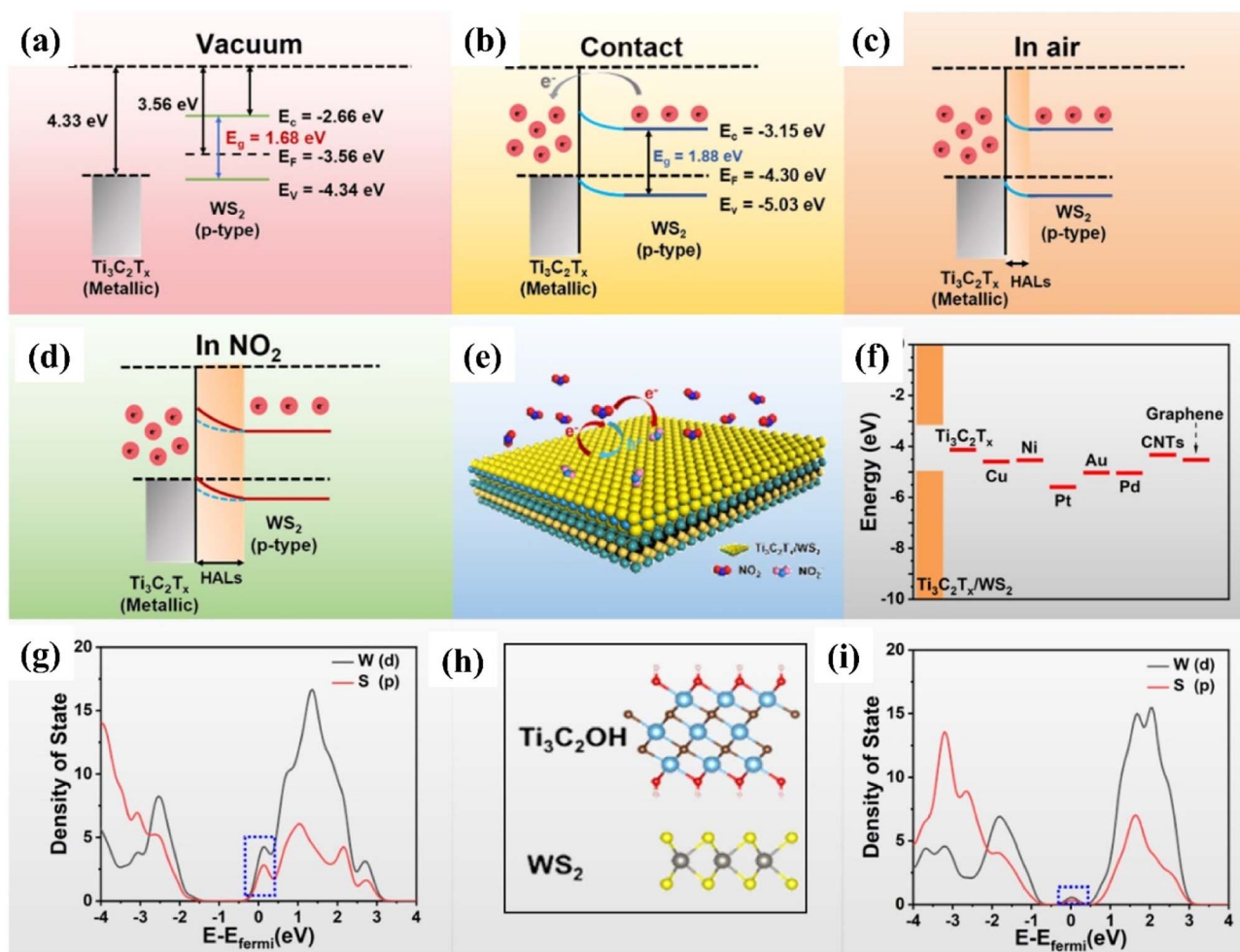


Fig. 8 Schematic representation showing the Fermi level position and work function for the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> (MXene)/and WS<sub>2</sub> (p-type) semiconductor (a) before contact (b) after contact (c) with air, and (d) NO<sub>2</sub> at room temperature. (e) Schematic diagram of the charge transfer process of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> (MXene)/WS<sub>2</sub> heterostructures in the presence of NO<sub>2</sub> gas at room temperature. (f) Different material work function. PDOS of the WS<sub>2</sub> in contact with the (g) Au and (h) Ti<sub>3</sub>C<sub>2</sub>OH, respectively, reprinted with permission from ref. 80 from Copyright (2023) American Chemical Society.



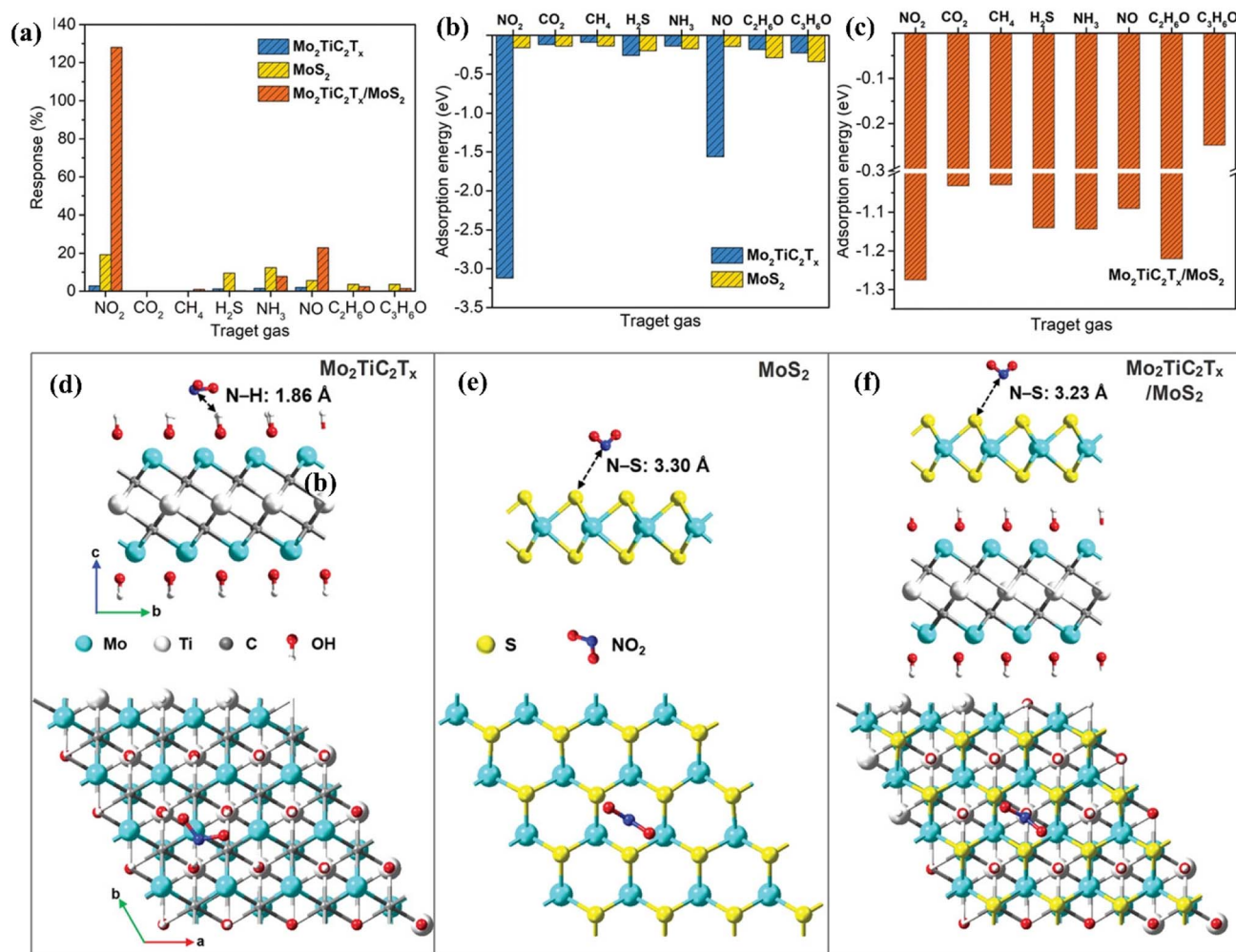


Fig. 9 (a) Selective responses for the Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub> (MXene), pristine MoS<sub>2</sub>, and Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub>/MoS<sub>2</sub>-based sensors in the presence of different gases. (b) Adsorption energies of pristine Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub> and pristine MoS<sub>2</sub> for different gases. (c) Adsorption energies of the Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub>/MoS<sub>2</sub> heterostructure for different gases. Top and side views of the configurations for (d) Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub> (e) MoS<sub>2</sub>, and (f) Mo<sub>2</sub>TiC<sub>2</sub>T<sub>x</sub>/MoS<sub>2</sub> heterostructure after adsorption of NO<sub>2</sub> molecules, reprinted in part permission from ref. 21 from Copyright (2023) Wiley materials.

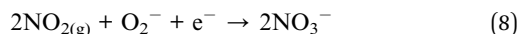
also been reported to have improved gas sensing performance with superior linearity, great limit of detection as low as 10 ppb, high selectivity, and response of 19.85% for 5 ppm NO<sub>2</sub> at ambient temperature.<sup>26</sup>

**3.2.2. MXene-TMD hybrids.** Due to the sharp conductivity change upon exposure to gas molecules, 2D/2D heterostructure materials are ideal candidates for gas sensors due to improved carrier transportation, synergistic effects, combined functionalities, spontaneous electron transfer, and formed a hetero-junction barrier at the interface, among other things.<sup>79,80</sup> These benefits led to the exploration of several 2D transition metal dichalcogenides (TMDs) and their MXene-based 2D/2D hybrids for NO<sub>2</sub> gas sensing applications. Due to their high mechanical characteristics, excellent response for redox reactions, huge number of active sites, outstanding adsorption capabilities, and tuneable layer-dependent features, 2D TMDs have recently attracted interest for gas sensing applications.<sup>81,82</sup> Among all the TMDs, WS<sub>2</sub> and its hybrids have emerged as the most promising candidate for NO<sub>2</sub> gas sensing application because of their

several advantages such as tuneable band structure, low cost, high surface area, electron mobility (234 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>) and ambipolar field modulation behaviour.<sup>83,84</sup> The target gases such as NO<sub>2</sub> can easily diffuse between the layers of WS<sub>2</sub>-MXene hybrids because W-S atoms in the 2D WS<sub>2</sub> are covalently bonded with weak van der Waals forces between the layers. Quan *et al.* for the first time reported a paper-based NO<sub>2</sub> sensor using the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/WS<sub>2</sub> heterostructure whose response (15.2%) was 3.2 and 76 times higher than that of the Au interdigital electrode integrated with the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/WS<sub>2</sub> sensor (4.8%) and Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> sensor (0.2%), respectively.<sup>85</sup> The sensor exhibited excellent stability even under high humid conditions with a limit of detection of 11 ppb for NO<sub>2</sub> gas. The NO<sub>2</sub> sensing process is explained in terms of band bending in hetero-junctions and changes in the width of the depletion layer upon exposure to gas molecules (Fig. 8). Free electrons migrate from WS<sub>2</sub> to Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> to balance the Fermi level position at equilibrium, and the adsorbed oxygen ions (O<sub>2</sub><sup>-</sup>) distribute across the material's surface, resulting in the production of a hole



accumulation layer. Because  $\text{NO}_2$  has a larger electron affinity (2.30 eV) than  $\text{O}_2$  (0.44 eV), electrons migrate towards  $\text{NO}_2$  from  $\text{NO}_3^-$  in the final step of the reaction. *i.e.*



DFT calculations confirmed that the improved  $\text{NO}_2$  sensing performance is credited to the work function matching, heterojunction regulation effect and suppression of the metal-induced gas states. The enhanced visible light photo-activation effects, optoelectronic properties along with the efficient separation of photo carriers by the 2D/2D heterointerface of  $\text{Ti}_3\text{C}_2\text{T}_x/\text{WS}_2$  helped to achieve enhanced  $\text{NO}_2$  sensing performance with full reversibility, good selectivity, fast response/recovery rate, long stability and low limit of detection (10 ppb).<sup>86</sup>

$\text{MoS}_2/\text{MXene}$  heterostructures with interconnected networks exhibited highly sensitive and selective  $\text{NO}_2$  sensing properties due to the presence of abundant Mo active sites, excellent heterointerface contacts and accelerated electrons from the

conductive MXene.<sup>18,19</sup> The excellent response (65.6%) of the 2H  $\text{MoS}_2/\text{Ti}_3\text{C}_2\text{T}_x$  MXene heterostructure to 100 ppm  $\text{NO}_2$  at ambient temperature is ascribed to the quick channels for carrier transportation and a large number of active sites between 2H  $\text{MoS}_2$  and few layered MXenes.<sup>19</sup> For a 2D/2D/2D composite made up of  $\text{Ti}_3\text{C}_2\text{T}_x$  MXene@ $\text{TiO}_2$ @ $\text{MoS}_2$ , the strong interfacial contact between the different components facilitated the charge carrier transfer and spatial separation, resulting in improved sensing performance, with  $\text{Ti}_3\text{C}_2\text{T}_x$  and  $\text{MoS}_2$  acting as the electron reservoir and main sensitive materials, respectively.<sup>20</sup> This  $\text{NO}_2$  sensor based on  $\text{Ti}_3\text{C}_2\text{T}_x$  MXene@ $\text{TiO}_2$ @ $\text{MoS}_2$  exhibited good response ( $R_a/R_g = 55.6$  for 50 ppm  $\text{NO}_2$ ) which was 3.8, 7.3 and 2.1 times higher than that of  $\text{TiO}_2$ @ $\text{MoS}_2$ , pristine  $\text{MoS}_2$  and  $\text{MXene}@$  $\text{MoS}_2$  composites respectively. The highly active double transition metal titanium molybdenum carbide ( $\text{Mo}_2\text{TiC}_2\text{T}_x$ ) and its hybrids with  $\text{MoS}_2$  displayed exceptional responsiveness due to their extremely strong surface adsorption ( $-3.12$  eV) for the gas  $\text{NO}_2$ .<sup>21</sup> The fabricated  $\text{Mo}_2\text{TiC}_2\text{T}_x/\text{MoS}_2$  sensor exhibited a sensitivity of around  $7.36\%$  ppm<sup>-1</sup>, detection limit of 2.5 ppm and room temperature reversibility (Fig. 9). The edge-enriched  $\text{Mo}_2\text{TiC}_2\text{T}_x/$

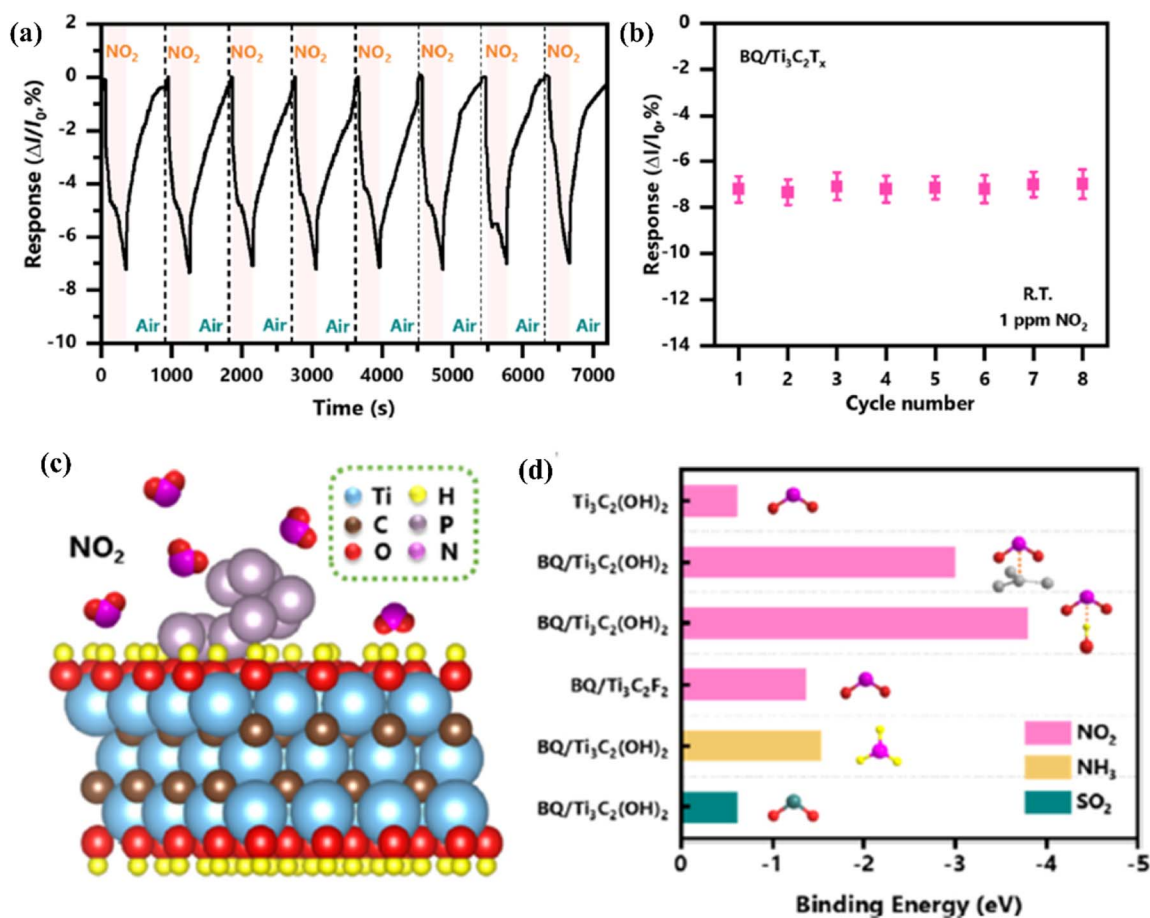


Fig. 10 Sensing properties of  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) modified with black phosphorus quantum dots in the presence of  $\text{NO}_2$  (a) cycling test and (b) the summarized responses of the  $\text{BQ}/\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) sensor to  $\text{NO}_2$  (1 ppm). (c) Schematic representation of adsorption of  $\text{NO}_2$  molecules on the  $\text{BQ}/\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) heterostructure surface. (d) Binding energies calculated of gas molecules on the pristine  $\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) and  $\text{BQ}/\text{Ti}_3\text{C}_2\text{T}_x$  (MXene) heterostructure with different binding modes and functional groups, reprinted with permission from ref. 85 from Copyright (2023) Elsevier.





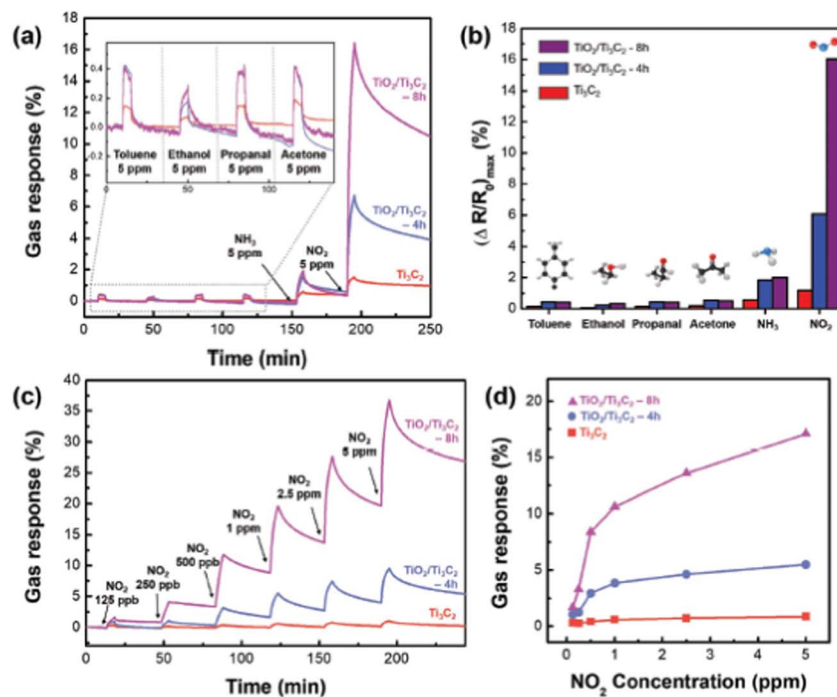


Fig. 12 (a) Selective response to gas and (b) maximum change in resistance when exposed to 5 ppm of toluene, ethanol, propanol, acetone, NH<sub>3</sub><sup>+</sup>, and NO<sub>2</sub>. (c) The TiO<sub>2</sub>/Ti<sub>3</sub>C<sub>2</sub> real-time gas response curve as a function of NO<sub>2</sub> concentration and (d) graph showing maximal resistance change at room temperature (NO<sub>2</sub>, at concentrations ranging from 0.125 to 5 ppm), reprinted in part permission from ref. 93 from Copyright (2023) Wiley materials.

APTES MXene-based gas sensor exhibited a good sensing response (2.5 times greater than pristine MXene) with stability for more than 45 days. Owing to its electronegativity, abundant functional groups and blocking effect,  $\gamma$ -poly ( $\gamma$ -glutamic acid) ( $\gamma$ -PGA) is used to stimulate the positive response behaviour of the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene-based sensor.<sup>96</sup> The sensing performance of MXene/ $\gamma$ -PGA-based sensors was improved by 85 times (1127.3%) that of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene (13.2%). Further, the sensor exhibited a faster response/recovery time (43.4 s/3 s) compared to the pristine MXene-based sensor (18.5 min/18.3 min) with excellent reversibility and repeatability. Fig. 11 shows the schematic illustration of the fabrication process and the sensing mechanisms of the MXene/ $\gamma$ -PGA sensor. Hassan *et al.* used embedded MXene-PEDOT-PSS as a reverse side layer and as the Joule heater for the printed CNT-graphene-based NO<sub>2</sub> sensor which was capable of detecting 1 ppm NO<sub>2</sub> gas at 65 °C with high selectivity.<sup>91</sup>

### 3.3. MXene-derived materials

Ti<sub>3</sub>C<sub>2</sub> MXene-derived semiconducting-metallic Ti<sub>3</sub>C<sub>2</sub>-TiO<sub>2</sub> hybrid materials are reported to show selective NO<sub>2</sub> sensing properties due to the analyte surface charge transfer and the modulation of Schottky barrier (SB) at the interface between the semiconducting and metallic surfaces.<sup>97</sup> The TiO<sub>2</sub>/Ti<sub>3</sub>C<sub>2</sub> composite-based sensors showed selective NO<sub>2</sub> sensing performance with excellent sensitivity around 13.7 times greater than pristine Ti<sub>3</sub>C<sub>2</sub> MXene and a limit of detection of 125 ppb (Fig. 12). Similarly, in another study, Liu *et al.* reported that the

composite with an optimal ratio of Ti<sub>3</sub>C<sub>2</sub>-TiO<sub>2</sub> showed higher response values (86 times) and faster recovery and response times (3.8 and 2 times) to 100 ppm NO<sub>2</sub> as compared to the pristine Ti<sub>3</sub>C<sub>2</sub> MXene.<sup>98</sup> Theoretical calculations proved that the two Ti-O bonds and the development of a Ti-N bond between the N and O atoms from the NO<sub>2</sub> and the nearby Ti atoms from the Ti<sub>3</sub>C<sub>2</sub> in the composites result in a substantial rise in the adsorption energy. Song *et al.* reported a high-performance NO<sub>2</sub> sensor based on MXene-derived TiO<sub>2</sub> nanoparticle intercalated between reduced graphene oxide (rGO) assembly in which uniform distribution of the TiO<sub>2</sub> nanoparticles and highly wrinkled rGO interconnected porous structure contributed to the sensing.<sup>99</sup> The MXene-derived TiO<sub>2</sub> spaced rGO gas sensor exhibited a 400% enhancement in NO<sub>2</sub> sensitivity with a limit of detection of around 50 ppb, excellent workability under humid conditions and good selectivity. Further, SnS<sub>2</sub>-MXene-derived TiO<sub>2</sub> hybrid materials exhibited a large response of 115 against 1000 ppm NO<sub>2</sub> gas with an ultrafast recovery time of 10 s at room temperature.<sup>100</sup>

## 4. Conclusions and future directions

In this review article, we highlighted the state-of-the-art use of MXene-based materials for room temperature NO<sub>2</sub> sensing applications. The advantages of MXenes such as their high surface area to volume ratio, tuneable physicochemical properties, high metallic characteristics, hydrophilicity, mechanical flexibility and availability of abundant surface functional groups make them ideal candidates for the fabrication of high





performance NO<sub>2</sub> sensor applications. Subsequently, we discussed the NO<sub>2</sub> sensing mechanisms of pristine and heterostructures of MXenes in detail. Among the various reported MXenes, Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> is the most explored one for NH<sub>3</sub> and VOC sensing. However, it possesses small adsorption energy (>−0.8 eV) for NO<sub>2</sub> gas suggesting its limited selectivity. The strong interlayer van der Waals force of attraction lead to the self-stacking of MXene nanosheets leading to the obstruction in the diffusion pathways and insufficient utilization of surface active sites, thereby reducing gas-sensing response. NO<sub>2</sub> molecules capture electrons from the Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> to produce NO<sub>2</sub><sup>−</sup> and oxidize Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> that leads to poor reversibility. Due to these demerits, the sensing performance of other MXenes such as Mo<sub>2</sub>CT<sub>x</sub> and V<sub>2</sub>CT<sub>x</sub> are also investigated, and it was discovered that these MXenes have great potential in NO<sub>2</sub> gas sensing applications, which opens up the door for unexplored MXenes for NO<sub>2</sub> sensing applications. The literature studies prove that the metallic conductivity of MXenes along with interlayer spacing is responsible for positive change in resistance.

The strategies such as introducing interlayer spacers (*e.g.*, TiO<sub>2</sub>), constructing self-supporting architectures and hybrid formation with other semiconducting materials are adopted to overcome the self-restacking problem of MXenes and thereby enhancing the adsorption site. From the literature, we can see that MXene hybrid formation with different metal oxides, TMDs, black phosphorus, polymers, *etc.* among others helps in the fabrication of MXene-based high-performance NO<sub>2</sub> sensors with high response, selectivity, low response and recovery time, *etc.* The sensor's response value and response speed are improved here due to the sufficient and compact interface contact, which can promote interfacial charge transfer. Furthermore, the *in situ* formed heterogeneous composite shows potential in gas sensor applications. This *in situ* heterogeneous composite production incorporates the structural benefits of multi-component materials while preventing the self-stacking of MXene nanosheets.

However, although there are reports on these materials for NO<sub>2</sub> sensing still there is plenty of room in this topic which need to be explored for the design of high performance NO<sub>2</sub> sensors. At this point, research on monolayer MXenes in gas sensors is in its early stage, despite the fact that Choi *et al.* revealed that monolayer Mo<sub>2</sub>CT<sub>x</sub> performs better against NO<sub>2</sub> but performs poorly without intercalants. Thus, the sensing capability of monolayer MXenes falls short of the criteria for practical applications, specifically in terms of its sensitivity and low applicability under ambient conditions.<sup>53</sup> This can be clearly seen from Fig. 13, where the pristine MXene shows less response towards NO<sub>2</sub> and its modifications with polymers help to enhance its sensitivity towards NO<sub>2</sub>. Several newly reported MXenes are yet to be explored for the fabrication of high performance NO<sub>2</sub> sensor devices.<sup>25</sup> Similarly, heterostructures of these newly emerging MXenes with other 2D materials such as TMDs, BP, MBenes, *etc.* are less explored. Hence, choosing the appropriate combination of materials with MXenes for the fabrication of heterostructures and tuning their properties by defect and vacancy engineering, alloying, doping, intercalation,

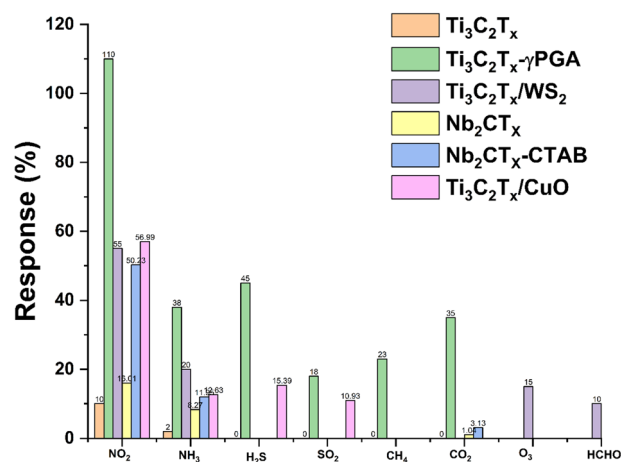


Fig. 13 Selective nature of MXenes towards NO<sub>2</sub> as compared to other gases.

layer tuning, *etc.* can allow for high performance selective NO<sub>2</sub> gas sensing performance.

Aside from sensor recovery and response time, the immediate response of the gas sensor is an important aspect. The response time of each sensor is determined by how quickly the gas molecules react to the sensing film and change their corresponding parameter. So far, the observed response time of NO<sub>2</sub> molecule detection by MXene's has been in the few seconds' range. As a result, developing NO<sub>2</sub> sensors capable of responding in milliseconds or microseconds remains difficult, and only one study is now available.<sup>69</sup> The technique for improving ultrafast sensors is based mostly on the interaction of gas molecules and MXenes as well as charge transfer in MXenes. By developing MXene-based heterostructures as sensing devices, the rapid charge carrier separation can be increased. Different methods such as photoexcitation, doping, gating, defect and vacancy engineering, surface modification, piezotronic/piezophotonic effects, *etc.* for the MXene-based NO<sub>2</sub> sensor devices are yet to be explored.

Recently, it is revealed that 2D material-based gas sensors with Schottky contact can create highly selective and sensitive sensors by adjusting the Schottky barrier height (SBH), which works as the gate controller to regulate the current flow. As a result of the metallic character of MXenes, SBH-controlled high-performance NO<sub>2</sub> selective sensors can be produced by selecting appropriate semiconducting materials for Schottky contact creation.<sup>101–104</sup> It is also necessary to strengthen theoretical and experimental efforts with thorough insight and understanding, which will lead to the advancement of high-performance NO<sub>2</sub> sensors. Several pathways for developing high-performing electrical contacts also should be identified.

Environmental factors include contaminants of different chemicals, humidity, moisture, corrosion caused by toxic vapours, and residual charges all have a substantial impact on a film's conductivity. These factors significantly lower the gas sensors' stability, reliability and repeatability. The response of MXene-based RT NO<sub>2</sub> sensors has been found to decrease with an increase in humidity. This is due to the interaction between



adsorbed oxygen  $O_2^-$  and water molecules that led to decreasing the site required for the adsorption of  $NO_2$  molecules.<sup>68</sup> So, various approaches for reducing humidity interference such as (i) pre-treating the target gas such as  $NO_2$  using dehumidifier (ii) modifying materials using hydrophobic materials and (iii) establishing a humidity compensation model are frequently utilized. Zhao *et al.* demonstrated the humidity compensation model can be established for  $NO_2$  gas sensing at a concentration of 2–10 ppm. They also applied statistical regression to find the relationship between gas concentration, humidity and gas sensing response ( $R$ ). The sensor based on MXene/ $\gamma$ -PGA was able to recognize  $NO_2$  with 2 ppm concentration after humidity compensations. Thus, it is necessary to make efforts to improve the sensing devices' stability and response.<sup>92</sup>

According to WHO, the recommended levels of  $NO_2$  exposure for an hour are 82 ppb and for a year are 410 ppb. Long-term exposure to  $NO_2$  above that threshold has negative health effects. The MXene-based  $NO_2$  sensor's lower detection limit has been measured in ppb. Consequently, it takes a lot of work to produce ultrasensitive  $NO_2$  sensors, which is an important task. Finding  $NO_2$ -sensitive materials that can quickly and easily integrate with MXenes and quickly detect  $NO_2$  at lower concentrations is crucial. Additionally, for quick sensor response, such materials need to speed up the transfer of charges.

Also, considering the advantages of 2D MXenes, flexible and wearable sensor devices based on these materials need to be explored. Till now there are only two reports on self-powered

$NO_2$  sensors based on MXenes which displayed promising features over conventional gas sensor devices. Hence, this research topic is believed to be an emerging research area in recent years.

Spectroscopic techniques that use electrical shields and laser sources have recently caught the attention of scientists for  $NO_2$  detection at trace quantity. The visible range of the absorption spectrum of  $NO_2$  molecules provides a significant opportunity for electronic exciton in  $NO_2$  molecules. Using spectroscopic methods for  $NO_2$  trace detection with MXene sensors could be a novel strategy. Over the past two years, the scientific community has become interested in light-assisted sensing of  $NO_2$  gas molecules. The MXene's ability to combine with metal layers has gained a lot of attention in surface plasma resonance (SPR) sensors.<sup>105</sup> Thus, the SPR properties of MXenes may be an unconventional approach to construct  $NO_2$  gas sensors based on MXenes. New experimental attempts should be directed towards realising the potential of plasmonic in the field of gas sensing. SPR can activate the interface between MXene and metal, changing the refractive index. Thus, a good selection of metal NPs and appropriate wavelengths will aid in the development of high-performance  $NO_2$  gas sensors.

By considering these aspects, it can be concluded that MXenes are still in the initial phase of gas sensor research and further exploration into these topics is needed to achieve high performance  $NO_2$  sensor devices (Fig. 14). Understanding the  $NO_2$  gas sensing mechanisms of MXene-based materials by different *in situ/operando* spectroscopic studies and by detailed theoretical investigations is of significant importance since it is

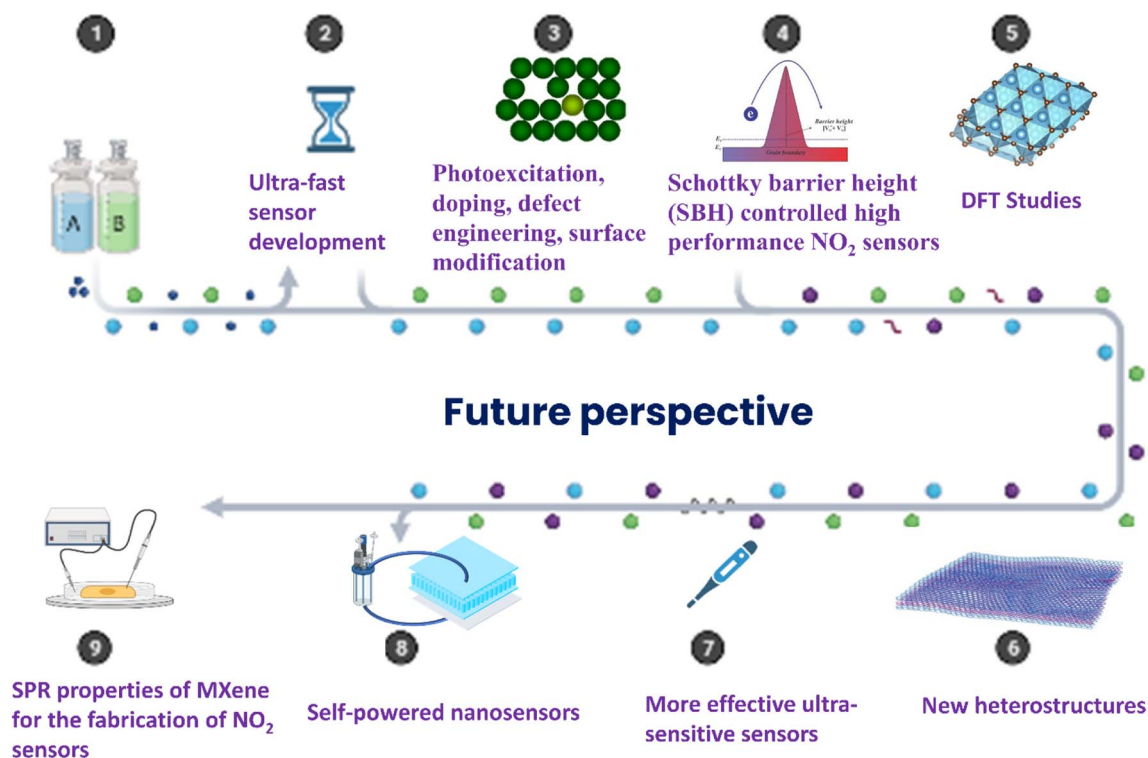


Fig. 14 Future perspectives of MXene-based  $NO_2$  gas sensors, reprinted with permission from, *Sci. Rep.*, 9, <https://doi.org/10.1038/s41598-019-45162-7> and <http://biorender.com/>.



helpful to design high performance gas sensors.<sup>37,106–109</sup> These studies can provide detailed information on the physical and chemical properties of the MXenes upon interaction with the analyte gas molecules. But there is no report on these topics to date. As a result, these study domains must be investigated in the next few years in order to develop superior gas sensor devices.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 L. Calderón-Garcidueñas, B. Azzarelli, H. Acuna, R. Garcia, T. M. Gambling, N. Osnaya, S. Monroy, M. Del Rosario Tizapantzi, J. L. Carson, A. Villarreal-Calderon and B. Rewcastle, *Toxicol. Pathol.*, 2002, **30**, 373–389.
- 2 N. M. Elsayed, *Toxicology*, 1994, **89**, 161–174.
- 3 J. A. Bernstein, N. Alexis, H. Bacchus, I. L. Bernstein, P. Fritz, E. Horner, N. Li, S. Mason, A. Nel, J. Oullette, K. Reijula, T. Reponen, J. Seltzer, A. Smith and S. M. Tarlo, *J. Allergy Clin. Immunol.*, 2008, **121**, 585–591.
- 4 D. S. Reay, E. A. Davidson, K. A. Smith, P. Smith, J. M. Melillo, F. Dentener and P. J. Crutzen, *Nat. Clim. Change*, 2012, **2**, 410–416.
- 5 A. R. Ravishankara, J. S. Daniel and R. W. Portmann, *Science*, 2009, **326**, 123–125.
- 6 J. N. Galloway, A. R. Townsend, J. W. Erisman, M. Bekunda, Z. Cai, J. R. Freney, L. A. Martinelli, S. P. Seitzinger and M. A. Sutton, *Science*, 2008, **320**, 889–892.
- 7 J. Ryu, S. Shim, J. Song, J. Park, H. S. Kim, S.-K. Lee, J. C. Shin, J. Mun and S.-W. Kang, *Nanomaterials*, 2023, **13**, 573.
- 8 G. Lei, C. Lou, X. Liu, J. Xie, Z. Li, W. Zheng and J. Zhang, *Sens. Actuators, B*, 2021, **341**, 129996.
- 9 M. Mathew and C. S. Rout, *J. Mater. Chem. C*, 2021, **9**, 395–416.
- 10 C. Jiang, G. Zhang, Y. Wu, L. Li and K. Shi, *CrystEngComm*, 2012, **14**, 2739–2747.
- 11 S. Liu, Z. Wang, Y. Zhang, Z. Dong and T. Zhang, *RSC Adv.*, 2015, **5**, 91760–91765.
- 12 S. Srivastava, K. Jain, V. N. Singh, S. Singh, N. Vijayan, N. Dilawar, G. Gupta and T. D. Senguttuvan, *Nanotechnology*, 2012, **23**, 205501.
- 13 T. Hyodo, K. Sasahara, Y. Shimizu and M. Egashira, *Sens. Actuators, B*, 2005, **106**, 580–590.
- 14 S. Liu, M. Wang, C. Ge, X. Zhang, S. Lei, S. Hussain, M. Wang, G. Qiao and G. Liu, *Appl. Surf. Sci.*, 2023, **610**, 155440.
- 15 T. Xie, G. Xie, H. Du, Y. Zhou, F. Xie, Y. Jiang and H. Tai, *IEEE Sens. J.*, 2016, **16**, 1865–1871.
- 16 A. Lv, Y. Pan and L. Chi, *Sensors*, 2017, **17**, 213.
- 17 W. Yuan, L. Huang, Q. Zhou and G. Shi, *ACS Appl. Mater. Interfaces*, 2014, **6**, 17003–17008.
- 18 V. T. Le, Y. Vasseghian, V. D. Doan, T. T. T. Nguyen, T.-T. Thi Vo, H. H. Do, K. B. Vu, Q. H. Vu, T. Dai Lam and V. A. Tran, *Chemosphere*, 2022, **291**, 133025.
- 19 Q. Thanh Hoai Ta, N. Ngoc Tri and J.-S. Noh, *Appl. Surf. Sci.*, 2022, **604**, 154624.
- 20 Z. Liu, H. Lv, Y. Xie, J. Wang, J. Fan, B. Sun, L. Jiang, Y. Zhang, R. Wang and K. Shi, *J. Mater. Chem. A*, 2022, **10**, 11980–11989.
- 21 Q. Zhao, W. Zhou, M. Zhang, Y. Wang, Z. Duan, C. Tan, B. Liu, F. Ouyang, Z. Yuan, H. Tai and Y. Jiang, *Adv. Funct. Mater.*, 2022, **32**, 2203528.
- 22 R. Frisenda, A. J. Molina-Mendoza, T. Mueller, A. Castellanos-Gomez and H. S. J. van der Zant, *Chem. Soc. Rev.*, 2018, **47**, 3339–3358.
- 23 H. Jin, C. Guo, X. Liu, J. Liu, A. Vasileff, Y. Jiao, Y. Zheng and S.-Z. Qiao, *Chem. Rev.*, 2018, **118**, 6337–6408.
- 24 X. Zhou, X. Hu, J. Yu, S. Liu, Z. Shu, Q. Zhang, H. Li, Y. Ma, H. Xu and T. Zhai, *Adv. Funct. Mater.*, 2018, **28**, 1706587.
- 25 A. V. Agrawal, N. Kumar and M. Kumar, *Nano-Micro Lett.*, 2021, **13**, 38.
- 26 Z. Yang, H. Zou, Y. Zhang, F. Liu, J. Wang, S. Lv, L. Jiang, C. Wang, X. Yan, P. Sun, L. Zhang, Y. Duan and G. Lu, *Adv. Funct. Mater.*, 2022, **32**, 2108959.
- 27 K. A. Sree Raj, N. Barman, S. Radhakrishnan, R. Thapa and C. S. Rout, *J. Mater. Chem. A*, 2022, **10**, 23590–23602.
- 28 S. A. Thomas, A. Patra, B. M. Al-Shehri, M. Selvaraj, A. Aravind and C. S. Rout, *J. Energy Storage*, 2022, **55**, 105765.
- 29 V. Chaudhary, N. Ashraf, M. Khalid, R. Walvekar, Y. Yang, A. Kaushik and Y. K. Mishra, *Adv. Funct. Mater.*, 2022, **32**, 2112913.
- 30 M. S. Bhargava Reddy, S. Kailasa, B. C. G. Marupalli, K. K. Sadasivuni and S. Aich, *ACS Sens.*, 2022, **7**, 2132–2163.
- 31 S. Mehdi Aghaei, A. Aasi and B. Panchapakesan, *ACS Omega*, 2021, **6**, 2450–2461.
- 32 R. Bhardwaj and A. Hazra, *J. Mater. Chem. C*, 2021, **9**, 15735–15754.
- 33 Y. Jian, D. Qu, L. Guo, Y. Zhu, C. Su, H. Feng, G. Zhang, J. Zhang, W. Wu and M.-S. Yao, *Front. Chem. Sci. Eng.*, 2021, **15**, 505–517.
- 34 C. Hu, X. Yu, Y. Li, J. Cheng, Q. Li and B. Xiao, *Appl. Surf. Sci.*, 2022, **592**, 153296.
- 35 A. V. Agrawal, N. Kumar and M. Kumar, *Nano-Micro Lett.*, 2021, **13**, 38.
- 36 M. Mathew, P. V. Shinde, R. Samal and C. S. Rout, *J. Mater. Sci.*, 2021, **56**, 9575–9604.
- 37 A. Sharma and C. S. Rout, *J. Mater. Chem. A*, 2021, **9**, 18175–18207.



- 38 Q. Yue, Z. Shao, S. Chang and J. Li, *Nanoscale Res. Lett.*, 2013, **8**, 425.
- 39 S. Nahiriak and B. Saruhan, *Sensors*, 2022, **22**, 972.
- 40 E. Lee, A. VahidMohammadi, B. C. Prorok, Y. S. Yoon, M. Beidaghi and D.-J. Kim, *ACS Appl. Mater. Interfaces*, 2017, **9**, 37184–37190.
- 41 S. J. Kim, H.-J. Koh, C. E. Ren, O. Kwon, K. Maleski, S.-Y. Cho, B. Anasori, C.-K. Kim, Y.-K. Choi, J. Kim, Y. Gogotsi and H.-T. Jung, *ACS Nano*, 2018, **12**, 986–993.
- 42 M. Wu, M. He, Q. Hu, Q. Wu, G. Sun, L. Xie, Z. Zhang, Z. Zhu and A. Zhou, *ACS Sens.*, 2019, **4**, 2763–2770.
- 43 H.-J. Koh, S. J. Kim, K. Maleski, S.-Y. Cho, Y.-J. Kim, C. W. Ahn, Y. Gogotsi and H.-T. Jung, *ACS Sens.*, 2019, **4**, 1365–1372.
- 44 Y. Zhang, Y. Jiang, Z. Duan, Q. Huang, Y. Wu, B. Liu, Q. Zhao, S. Wang, Z. Yuan and H. Tai, *Sens. Actuators, B*, 2021, **344**, 130150.
- 45 W.-T. Koo, H.-J. Cho, D.-H. Kim, Y. H. Kim, H. Shin, R. M. Penner and I.-D. Kim, *ACS Nano*, 2020, **14**, 14284–14322.
- 46 H. Hashtroudi, I. D. R. Mackinnon and M. Shafiei, *J. Mater. Chem. C*, 2020, **8**, 13108–13126.
- 47 R. Malik, V. K. Tomer, Y. K. Mishra and L. Lin, *Applied Physics Reviews*, 2020, **7**, 021301.
- 48 Y. Han, D. Huang, Y. Ma, G. He, J. Hu, J. Zhang, N. Hu, Y. Su, Z. Zhou, Y. Zhang and Z. Yang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 22640–22649.
- 49 S. Gasso and A. Mahajan, *Mater. Sci. Semicond. Process.*, 2022, **152**, 107048.
- 50 J. Choi, *Adv. Funct. Mater.*, 2020, 2003998.
- 51 Q. Zhao, D. Sun, S. Wang, Z. Duan, Z. Yuan, G. Wei, J.-L. Xu, H. Tai and Y. Jiang, *ACS Sens.*, 2021, **6**, 2858–2867.
- 52 Z. Yang, A. Liu, C. Wang, F. Liu, J. He, S. Li, J. Wang, R. You, X. Yan, P. Sun, Y. Duan and G. Lu, *ACS Sens.*, 2019, **4**, 1261–1269.
- 53 J. Choi, B. Chacon, H. Park, K. Hantanasirisakul, T. Kim, K. Shevchuk, J. Lee, H. Kang, S.-Y. Cho, J. Kim, Y. Gogotsi, S. J. Kim and H.-T. Jung, *ACS Sens.*, 2022, **7**, 2225–2234.
- 54 Y. Wang, J. Fu, J. Xu, H. Hu and D. Ho, *ACS Appl. Mater. Interfaces*, 2023, **15**, 12232–12239.
- 55 M. Hu, C. Cui, C. Shi, Z.-S. Wu, J. Yang, R. Cheng, T. Guang, H. Wang, H. Lu and X. Wang, *ACS Nano*, 2019, **13**, 6899–6905.
- 56 K. Rathi, N. K. Arkoti and K. Pal, *Adv. Mater. Interfaces*, 2022, **9**, 2200415.
- 57 X. Yu, Y. Li, J. Cheng, Z. Liu, Q. Li, W. Li, X. Yang and B. Xiao, *ACS Appl. Mater. Interfaces*, 2015, **7**, 13707–13713.
- 58 Q. Guo, W. Pang, X. Xie, Y. Xu and W. Yuan, *J. Mater. Chem. A*, 2022, **10**, 15634–15646.
- 59 S.-Y. Cho, J. Y. Kim, O. Kwon, J. Kim and H.-T. Jung, *J. Mater. Chem. A*, 2018, **6**, 23408–23416.
- 60 Y. Zhang, Y. Jiang, Z. Duan, Q. Huang, Y. Wu, B. Liu, Q. Zhao, S. Wang, Z. Yuan and H. Tai, *Sens. Actuators, B*, 2021, **344**, 130150.
- 61 A. Lipatov, M. Alhabeib, M. R. Lukatskaya, A. Boson, Y. Gogotsi and A. Sinitskii, *Adv. Electron. Mater.*, 2016, **2**, 1600255.
- 62 K. Hantanasirisakul and Y. Gogotsi, *Adv. Mater.*, 2018, **30**, 1804779.
- 63 T. Schultz, N. C. Frey, K. Hantanasirisakul, S. Park, S. J. May, V. B. Shenoy, Y. Gogotsi and N. Koch, *Chem. Mater.*, 2019, **31**, 6590.
- 64 Y. Chae, S. J. Kim, S. Y. Cho, J. Choi, K. Maleski, B. J. Lee, H. T. Jung, Y. Gogotsi, Y. Lee and C. W. Ahn, *Nanoscale*, 2019, **11**(17), 8387–8393.
- 65 A. Naveen Kumar and K. Pal, *Mater. Adv.*, 2022, **3**, 5151–5162.
- 66 S. Liu, M. Wang, C. Ge, S. Lei, S. Hussain, M. Wang, G. Qiao and G. Liu, *Sens. Actuators, B*, 2022, **365**, 131919.
- 67 S. Kang, A. Mirzaei, K. Y. Shin, W. Oum, D. J. Yu, S. S. Kim and H. W. Kim, *Sens. Actuators, B*, 2023, **375**, 132882.
- 68 S. Gasso, M. K. Sohal and A. Mahajan, *Sens. Actuators, B*, 2022, **357**, 131427.
- 69 S. Gasso and A. Mahajan, *ACS Appl. Nano Mater.*, 2023, **6**(8), 6678–6692.
- 70 J. Wang, Y. Yang and Y. Xia, *Sens. Actuators, B*, 2022, **353**, 131087.
- 71 Z. Yang, L. Jiang, J. Wang, F. Liu, J. He, A. Liu, S. Lv, R. You, X. Yan, P. Sun, C. Wang, Y. Duan and G. Lu, *Sens. Actuators, B*, 2021, **326**, 128828.
- 72 X. Liu, H. Zhang, Y. Song, T. Shen and J. Sun, *Sens. Actuators, B*, 2022, **367**, 132025.
- 73 C. Fan, J. Shi, Y. Zhang, W. Quan, X. Chen, J. Yang, M. Zeng, Z. Zhou, Y. Su, H. Wei and Z. Yang, *Nanoscale*, 2022, **14**, 3441–3451.
- 74 F. Guo, C. Feng, Z. Zhang, L. Zhang, C. Xu, C. Zhang, S. Lin, H. Wu, B. Zhang, A. Tabusi and Y. Huang, *Sens. Actuators, B*, 2023, **375**, 132885.
- 75 D. Wang, D. Zhang, J. Guo, Y. Hu, Y. Yang, T. Sun, H. Zhang and X. Liu, *Nano Energy*, 2021, **89**, 106410.
- 76 S. Gasso and A. Mahajan, *ACS Sens.*, 2022, **7**, 2454–2464.
- 77 J. Fan, J. Gao, H. Lv, L. Jiang, F. Qin, Y. Fan, B. Sun, J. Wang, M. Ikram and K. Shi, *J. Mater. Chem. A*, 2022, **10**, 25714–25724.
- 78 B. Sun, H. Lv, Z. Liu, J. Wang, X. Bai, Y. Zhang, J. Chen, K. Kan and K. Shi, *J. Mater. Chem. A*, 2021, **9**, 6335–6344.
- 79 E. Lee, Y. S. Yoon and D.-J. Kim, *ACS Sens.*, 2018, **3**, 2045–2060.
- 80 Y. Kim, S. Lee, J.-G. Song, K. Y. Ko, W. J. Woo, S. W. Lee, M. Park, H. Lee, Z. Lee, H. Choi, W.-H. Kim, J. Park and H. Kim, *Adv. Funct. Mater.*, 2020, **30**, 2003360.
- 81 W.-T. Koo, H.-J. Cho, D.-H. Kim, Y. H. Kim, H. Shin, R. M. Penner and I.-D. Kim, *ACS Nano*, 2020, **14**, 14284–14322.
- 82 S. Kumar, V. Pavelyev, P. Mishra, N. Tripathi, P. Sharma and F. Calle, *Mater. Sci. Semicond. Process.*, 2020, **107**, 104865.
- 83 G. Sanyal, A. Vaidyanathan, C. S. Rout and B. Chakraborty, *Mater. Today Commun.*, 2021, **28**, 102717.
- 84 D. Simon Patrick, P. Bharathi, M. Krishna Mohan, C. Muthamizhelvan, S. Harish and M. Navaneethan, *J. Mater. Sci.: Mater. Electron.*, 2022, **33**, 9235–9245.



- 85 W. Quan, J. Shi, H. Luo, C. Fan, W. Lv, X. Chen, M. Zeng, J. Yang, N. Hu, Y. Su, H. Wei and Z. Yang, *ACS Sens.*, 2023, **8**, 103–113.
- 86 Y. Xia, S. He, J. Wang, L. Zhou, J. Wang and S. Komarneni, *Chem. Commun.*, 2021, **57**, 9136–9139.
- 87 N. Mao, J. Tang, L. Xie, J. Wu, B. Han, J. Lin, S. Deng, W. Ji, H. Xu, K. Liu, L. Tong and J. Zhang, *J. Am. Chem. Soc.*, 2016, **138**, 300–305.
- 88 P. Yasaei, B. Kumar, T. Foroozan, C. Wang, M. Asadi, D. Tuschel, J. E. Indacochea, R. F. Klie and A. Salehi-Khojin, *Adv. Mater.*, 2015, **27**, 1887–1892.
- 89 Aaryashree, P. V. Shinde, A. Kumar, D. J. Late and C. S. Rout, *J. Mater. Chem. C*, 2021, **9**, 3773–3794.
- 90 Q. Xu, B. Zong, Y. Yang, Q. Li and S. Mao, *Sens. Actuators, B*, 2022, **373**, 132696.
- 91 K. Hassan, N. Stanley, T. T. Tung, P. L. Yap, H. Rastin, L. Yu and D. Losic, *Adv. Mater. Interfaces*, 2021, **8**, 2101175.
- 92 Q. Zhao, D. Sun, S. Wang, Z. Duan, Z. Yuan, G. Wei, J.-L. Xu, H. Tai and Y. Jiang, *ACS Sens.*, 2021, **6**, 2858–2867.
- 93 Y. Zhou, J. Qiu, H. Zhao, Y. Wang, J. Li and C. Zou, *J. Phys. Chem. Lett.*, 2022, **13**, 9599–9606.
- 94 H. Zhi, J. Gao and L. Feng, *ACS Sens.*, 2020, **5**, 772–780.
- 95 J. Wu, Z. Wu, W. Huang, X. Yang, Y. Liang, K. Tao, B.-R. Yang, W. Shi and X. Xie, *ACS Appl. Mater. Interfaces*, 2020, **12**, 52070–52081.
- 96 Q. Zhao, D. Sun, S. Wang, Z. Duan, Z. Yuan, G. Wei, J.-L. Xu, H. Tai and Y. Jiang, *ACS Sens.*, 2021, **6**, 2858–2867.
- 97 Q. T. H. Ta, N. M. Tran, N. N. Tri, A. Sreedhar and J.-S. Noh, *Chem. Eng. J.*, 2021, **425**, 131437.
- 98 S. Liu, M. Wang, G. Liu, N. Wan, C. Ge, S. Hussain, H. Meng, M. Wang and G. Qiao, *Appl. Surf. Sci.*, 2021, **567**, 150747.
- 99 Y. Song, Y. Xu, Q. Guo, Z. Hua, F. Yin and W. Yuan, *ACS Appl. Mater. Interfaces*, 2021, **13**, 39772–39780.
- 100 T. Chen, W. Yan, Y. Wang, J. Li, H. Hu and D. Ho, *J. Mater. Chem. C*, 2021, **9**, 7407–7416.
- 101 Ó. L. Camargo Moreira, W.-Y. Cheng, H.-R. Fuh, W.-C. Chien, W. Yan, H. Fei, H. Xu, D. Zhang, Y. Chen, Y. Zhao, Y. Lv, G. Wu, C. Lv, S. K. Arora, C. Ó Coileáin, C. Heng, C.-R. Chang and H.-C. Wu, *ACS Sens.*, 2019, **4**, 2546–2552.
- 102 A. Singh, Md. A. Uddin, T. Sudarshan and G. Koley, *Small*, 2014, **10**, 1555–1565.
- 103 Z. Yang, X. Dou, S. Zhang, L. Guo, B. Zu, Z. Wu and H. Zeng, *Adv. Funct. Mater.*, 2015, **25**, 4039–4048.
- 104 M. Mathew and C. S. Rout, *J. Mater. Chem. C*, 2021, **9**, 395–416.
- 105 V. K. Verma, R. Kumar, S. Pal and Y. K. Prajapati, *Opt. Mater.*, 2022, **133**, 112977.
- 106 T. Hartman, R. G. Geitenbeek, C. S. Wondergem, W. van der Stam and B. M. Weckhuysen, *ACS Nano*, 2020, **14**, 3725–3735.
- 107 S. Sänze, A. Gurlo and C. Hess, *Angew. Chem., Int. Ed.*, 2013, **52**, 3607–3610.
- 108 D. Degler, U. Weimar and N. Barsan, *ACS Sens.*, 2019, **4**, 2228–2249.
- 109 A. Bag and N.-E. Lee, *J. Mater. Chem. C*, 2019, **7**, 13367–13383.

