



Cite this: *Energy Adv.*, 2024,
3, 1367

Green synthesis of cobalt ferrite from rotten passion fruit juice and application as an electrocatalyst for the hydrogen evolution reaction†

Rochelin Prosper Medang,^a Roussin Lontio Fomekong,^b
Edwin Akongnwi Nforna,^c Hypolite Mathias Tedjiekeng Kamta,^b
Cédrik Ngnintedem Yonti,^b Patrice Kenfack Tsobnang,^d John Ngolui Lambi^b
and Dieudonné Bitondo^{a,e}

Green synthesis of an efficient and low-cost electrocatalyst for the hydrogen evolution reaction (HER) in a basic medium is highly desirable for renewable energy conversion. Herein, cobalt-ferrite (CoFe_2O_4) was prepared by green coprecipitation synthesis using a precipitating agent (citrate) extracted from rotten passion fruit. The preparation followed two steps, in which the cobalt-iron citrate precursor was first obtained by coprecipitation, followed by pyrolysis in the air at 850°C . While Fourier transform infrared (FTIR) and thermogravimetry (TG) confirmed the successful preparation of the precursor, X-ray diffraction (XRD), Raman spectroscopy and scanning electron microscopy (SEM) revealed without any doubt the formation of polyhedron-like shape CoFe_2O_4 . The electrocatalytic properties of the as-prepared material for the HER in 1 M KOH have been investigated. CoFe_2O_4 demonstrated good electrochemical performance with an overpotential of 440 mV at a current density of 10 mA cm^{-2} , a Tafel slope of 98 mV dec^{-1} , an electrical double layer capacitance (C_{dl}) of 0.43 mF cm^{-2} and very high stability with an overpotential loss of only 13.07 mV after 12 hours. This paper provides a simple, easy method for HER catalyst preparation by recycling waste fruit.

Received 13th September 2023,
Accepted 9th May 2024

DOI: 10.1039/d3ya00450c

rsc.li/energy-advances

1. Introduction

The conversion of protons into hydrogen from an inexpensive, stable and robust electrocatalyst has nowadays become a promising source of sustainable and carbon-free energy.^{1–4} The use of noble metals like platinum and palladium as appropriate electrocatalysts so far, however, has some

limitations, not only because of their rarity but also the requirement of costly operations for industrial implementation.^{5,6} Thus, the devolution of researchers towards the search for stable alternative electrocatalysts, at low cost and promoting large-scale applications, remains a current challenge.^{7,8} Very recent studies have revealed that transition metal ferrites in general and cobalt ferrite in particular, which can be obtained chemically in inexhaustible quantities, are prominently among the most efficient of these alternatives.^{9–11} The work of Hamza Belhadj *et al.* has elucidated the excellent electrocatalytic activity of some transition metal ferrites, foremost among which is cobalt ferrite (CoFe_2O_4), for the hydrogen evolution reaction (HER). Owing to its remarkable physicochemical properties, this material is perceived as an ideal candidate for the HER.¹ It belongs to the family of ferrites with the chemical formula MFe_2O_4 (where M is a divalent metal ion: Co^{2+} , Fe^{2+} , Zn^{2+}). It has an “inverse” spinel structure, with an ionic distribution as follows: $(\text{Fe}^{3+})_A(\text{Fe}^{3+}\text{Co}^{2+})_B\text{O}_4^{2-}$. With this structure, the oxygen ions whose ionic size is relatively larger, form a face-centered cubic network (CFC), defining tetrahedral A and octahedral B cationic sites. In this configuration, the Co^{2+} occupies the octahedral sites, hence the name reverse

^a Laboratoire Méthodes (LM), École Nationale Supérieure Polytechnique de Douala, Université de Douala, Douala, Cameroon

^b Department of Chemistry, Higher Teacher Training College, University of Yaoundé I, Yaoundé, Cameroon

^c Department of Fundamental Sciences, Higher Technical Teacher Training College, University of Bamenda, Bamili, Cameroon

^d Department of Chemistry, University of Dschang, P.O. Box 67, Dschang, Cameroon. E-mail: patrice.kenfack@univ-dschang.org

^e Département Qualité Industrielle, Hygiène, Sécurité Industrielle et Génie de l'Environnement, École Nationale Supérieure Polytechnique de Douala, Université de Douala, Douala, Cameroon

† Electronic supplementary information (ESI) available: FTIR and TGA spectra of Co- and Fe-precursors, PXRD pattern of the residue of Co-, Fe- and Co-Fe-precursors, SEM images of the Co_3O_4 and $\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4$, and cyclic voltammetric curves of CoFe_2O_4 at different scanning speeds. See DOI: <https://doi.org/10.1039/d3ya00450c>

The thermal behavior of the three precursors was studied by thermogravimetric analysis (TGA) using a Linseis STAPT-1000 device operating from room temperature to 1000 °C in air, at a heating rate of 10 °C min⁻¹. Powder X-ray diffraction (PXRD) measurements on the residues obtained after the thermal decomposition of the precursors were performed using a Siemens kristalloflex θ -2 θ D5000 powder diffractometer with the Bragg-Brentano geometry. These measurements were done in the 2 θ range of 10 to 90° steps size of 0.02° s⁻¹. Further evaluation of the material structure was performed using the Raman microscope (Renishaw, England). For this analysis, a 0.75 mW laser power and 532 nm laser with a 50× magnification objective were applied to the sample at different locations with an exposure time of 20 s between 100 and 900 cm⁻¹. The scanning electron microscope (SEM) brand Zeiss ultra 55 was used to determine the morphology and the average size of the particles of the materials. This microscope is equipped with an energy-dispersive X-ray (EDX) spectrometer from Oxford Instruments, Oxford, England. For this purpose, the experiments were carried out at 15 keV with a working distance (distance between the sample and the part of the lower pole in the SEM system) of 8 mm. A probe current of 1 nA and an acquisition time of 300 s were used to collect the chemical spectra.

2.5. Determination of the potential of CoFe₂O₄ for the electrocatalysis of the HER

A three-electrode system was used for the electrochemical HER measurements. All experiments were performed at room temperature on a PGSTAT 204 electrochemical workstation (Utrecht, the Netherlands, NOVA version 2.1.4) using a glassy carbon (GC) electrode coated with our material as the working electrode, Hg/HgO as the reference electrode (RE), and a carbon rod as the counter electrode. To prepare the working electrode, 20 μ L of the as-prepared suspension containing approximately 0.12 mg of CoFe₂O₄ was drop-cast onto the GC and dried under vacuum, leading to a catalyst loading of 1.7 mg cm⁻². Then, a drop of Nafion solution (5 wt%) was deposited on top and dried at 60 °C for 2 hours. The measured potentials were recalculated against a reversible hydrogen electrode (RHE) according to the Nernst equation: $E_{\text{RHE}} = E_{\text{Hg}}/\text{HgO} + 0.098 + 0.059\text{pH}$, where 0.098 V corresponds to the potential of the reference electrode at 0 pH. Linear sweep voltammetry (LSV) was measured in 1 M KOH (pH = 14) with a scan rate of 2 mV s⁻¹, and the overpotential was evaluated using the following equation $\eta = 0 - E$ (vs. RHE (V)). The Tafel slope (TS) was extracted using the equation $\eta = a + b \log|j|$, with η , a , b , and j corresponding to overpotential, Tafel constant, TS, and current density, respectively. Electrochemical impedance spectroscopy (EIS) was performed in the frequency range of 0.1–1000 Hz and the amplitude of the applied AC potential was 10 mV. The stability of the material was investigated by chronopotentiometry for 12 h. Cyclic voltammetry (CV) for the three samples at different scan rates was recorded in the potential range of (0.0–0.1 V). The electrical double layer capacitances (C_{dl}) were calculated from the recorded CV at a potential of 0.05 V, using the formula $C_{\text{dl}} = (J_{\text{a}} + |J_{\text{c}}|)/2\nu$, in which J_{a} and J_{c} are the anodic and cathodic

current density, respectively, recorded at 0.05 V and ν is the scan rate.

3. Results and discussion

3.1. Analysis of passion fruit juice

Table 1 presents the juice and fiber content of the passion fruits used in this work.

The dietary fiber content (8.93%) is within the range of values from 1.5% to 14.3% predicted by Favier.²⁷ This result reveals a non-negligible quantity of dietary fibers, which would have had an influence on the analysis results if they had not been eliminated by filtration during the extraction of the passion fruit juice.

Table S1 of the ESI† presents the volumes of NaOH solution (45 mL) obtained during the titration of the passion fruit juice used in this work.

The mass percentage (P') of citric acid in the solution obtained by using eqn (2) is 3.46%. This value is between 2.50% and 3.50% as predicted in the literature.²⁷ This result confirms that rotten passion fruit is a potential source of citric acid that can be used for the synthesis of oxide materials like CoFe₂O₄.

$$P' = \frac{C_a M V \times 100}{m_j} \quad (2)$$

(C_a : the molar concentration of the citric acid; M : the molar mass of the citric acid; V : the volume of the solution, m_j : mass of passion fruit juice).

3.2. Analyses of the carboxylate precursors

3.2.1. Fourier transformed infrared (FTIR) result. The infrared spectrum of the Co-Fe precursor is shown in Fig. 1 and those for the single metal Co-precursor and Fe-precursor are given in Fig. S1 and S2 (ESI†), respectively.

Table 2 presents the major vibration bands of some characteristic chemical groups of the synthesized compound.

Fig. 1 shows a broad absorption band at 3311 cm⁻¹, which can be attributed to the vibration of the O-H group associated with a sp³ carbon. The peaks at 2920 cm⁻¹ and 2850 cm⁻¹ ascribed to the CH₂ group correspond to $\nu_{\text{as}}(\text{CH}_2)$ and $\nu_{\text{s}}(\text{CH}_2)$ vibrations, respectively. They are very weak as expected and appear as the shoulder of the wide band of the O-H group. Indeed, the Rayleigh criterion, which states that “two lines of the same intensity are separated if the maximum of one corresponds to the first minimum of the other”,²⁹ is not

Table 1 Juice and fiber content in passion fruit

	Total number of fruits (sample): 13	
	Mass (g)	Mass percentage (%)
Passion fruit (m_T)	420.00 \pm 0.01	100.00
Passion fruit juice containing fiber (m_{jf})	109.30 \pm 0.01	26.02
Passion fruit juice (m_j)	71.80 \pm 0.01	17.09
Fibers (m_f)	37.50 \pm 0.01	8.93





Fig. 1 FTIR of the Co-Fe precursor.

Table 2 FTIR absorption bands of the Co-Fe precursor

Functional group	Experimental frequencies (cm ⁻¹)	Reference frequencies ²⁸
$\nu(\text{OH})$ associated	3311	3570–3200
$\nu_{\text{as}}(\text{COO}^-)$	1579	1610–1550
$\nu_{\text{as}}(\text{CH}_2)$	2920	2935–2915
$\nu_{\text{s}}(\text{CH}_2)$	2850	2865–2845
$\nu_{\text{s}}(\text{COO}^-)$	1386	1420–1300
(ter) $\nu_{\text{s}}\text{C-C}$	1052	~1000
$\delta(\text{M-O})$	381	~400

satisfied for these bands. Fig. 1 also shows absorption bands at 1579 cm⁻¹, 1386 cm⁻¹ and 1052 cm⁻¹, which respectively correspond to the asymmetric stretching of the CO bond in the COO⁻ group, the symmetric stretching of the CO bond in the same group, and the symmetric stretching of the C-C bond of the carbon skeleton. By comparing the IR spectra of citric acid³⁰ and that of the precursor, it can be seen that the peak at 1720 cm⁻¹ linked to CO in the COOH group observed in the spectrum of citric acid is shifted to 1579 cm⁻¹. This could be related to the fact that the carboxylate group interacts with the cobalt ion in the precursor *via* a coordination bond. So, this FTIR result confirms that a coordination compound with the citrate as a ligand has been synthesized as expected. The FTIR spectra of Co and Fe precursors are also shown in Fig. S1 and S2 of the ESI,[†] respectively. These figures show the appearance of similar absorption bands, corresponding to the same vibration frequencies of the chemical bonds found in the as-prepared compound treated in this article.

3.2.2. Thermal analysis of the precursors. Fig. 2 shows the TGA curves of the Co-Fe precursor and Fig. S3 and S4 of the ESI,[†] respectively the Co and Fe precursors.

The TGA curve of the Co-Fe precursor shows that its decomposition takes place in four steps. Stage 1 (mass loss of 9.53%) occurs between 50 °C and 180 °C (maximum reaction temperature of 174 °C) and corresponds to the loss of three

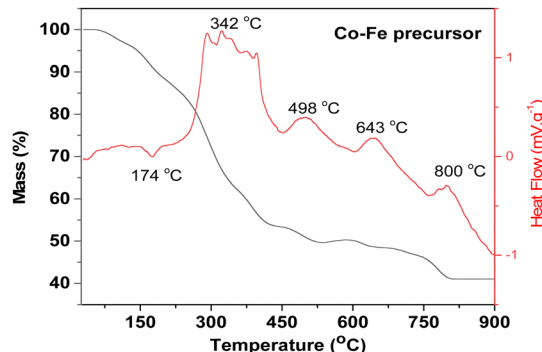


Fig. 2 Thermogravimetric (TGA) curve for the Co-Fe precursor.

water molecules with the theoretical percentage of 9.23% from the proposed Co-Fe formula of $\text{CoFe}_2(\text{C}_6\text{H}_5\text{O}_7) \cdot 2\text{H}_2\text{O}$. Step 2 (29.42% of experimental mass loss) occurs between 180 °C and 365 °C (maximum temperature of 342 °C). It can be attributed to the loss of four molecules of CO₂ (theoretical percentage of 30.08%). Step 3 (12.61% of experimental mass loss) occurs between 365 °C and 650 °C (maximum temperatures of 498 °C and 643 °C) and it can be attributed to the loss of three molecules of CO (theoretical percentage of 14.36%). The last stage (7.51% of experimental mass loss) occurs between 650 °C and 850 °C (maximum reaction temperature of 800 °C) and can be due to the loss of an organic molecule of CH₂CO (theoretical percentage of 7.18%). Beyond 850 °C, the residue of this thermal decomposition has an experimental percentage of 40.99%. This can be related to the formation of one mole of CoFe₂O₄ (theoretical percentage of 40.17% close to the experimental value). The same analysis done for the Co and Fe precursors (see Fig. S3 and S4 respectively, ESI,[†]) shows that their residues are Co₃O₄ and a mixture of Fe₂O₃ and Fe₃O₄, respectively.

3.3. Analyses of the synthesized oxides

3.3.1. The PXRD result of the decomposition residue. Fig. 3 shows the powder diffraction (PXRD) pattern of the residue after the thermal decomposition of the Co-Fe precursor and Fig. S5 and S6 of the ESI,[†] that of the Co and Fe-precursors.

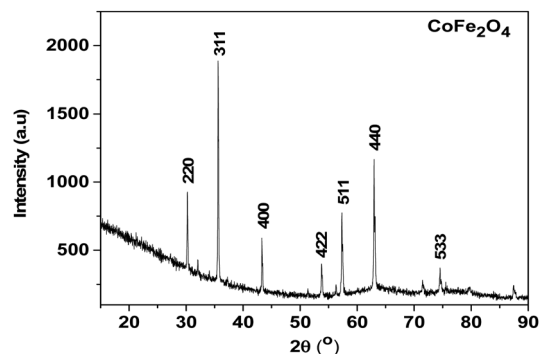


Fig. 3 X-ray powder diffraction pattern of the residue from the thermal decomposition of the Co-Fe precursor.



As shown in Fig. 3, almost all the peaks are well indexed with the CoFe_2O_4 structure (CDD reference of 1533163), and the very weak peaks that remain are those of the Fe_5LiO_8 structure (CDD reference 1541958). This material with a quantity less than 5 wt% would come from the few lithium ions that would have remained during the synthesis of the Co-Fe precursor synthesized from the rotten passion fruit in this work. Also, the peaks in Fig. S6 and S7 (ESI†) are well indexed, respectively, with the Co_3O_4 structure (COD reference 9005888) and the mixture of 63% of $\alpha\text{-Fe}_2\text{O}_3$ and 37% of Fe_3O_4 (COD references 2108027 and 9005814). All these results are in agreement with those of the thermogravimetry analysis shown above and they indicate that the juice obtained from the rotten passion fruit is a suitable reagent for the synthesis of these oxide materials, particularly CoFe_2O_4 for which its application for the HER is studied below. Such green synthesis of CoFe_2O_4 materials was reported in some previous works in the literature but to the best of our knowledge,^{31–33} this is the first time that rotten passion fruits were used for this purpose.

3.3.2. Raman spectrum of the residue from the thermal decomposition of the Co-Fe precursor. Fig. 4 shows the Raman spectrum of the thermal decomposition residue of the Co-Fe precursor and Fig. S7 and S8 of the ESI† show the decomposition residues of the Co and Fe-precursors, respectively.

Fig. 4 clearly shows peaks in the range of 200 to 750 cm^{-1} region. The peak that occurs at 202 cm^{-1} is attributed to the $\text{T}_{2g}(1)$ vibration mode corresponding to the translation motion of the whole tetrahedron in the CoFe_2O_4 material.³⁴ The peak at 300 cm^{-1} is attributable to the E_g vibration mode related to the symmetric bending of the $\text{Fe}(\text{Co})\text{-O}$ bond in agreement with the literature.³⁴ The peak at 485 cm^{-1} is attributable to the $\text{T}_{2g}(3)$ vibration mode of the asymmetric bending $\text{Fe}(\text{Co})\text{-O}$ bond. The peak at 611 cm^{-1} is attributed to the $\text{A}_{1g}(2)$ vibration mode of the symmetric stretching $\text{Fe}(\text{Co})\text{-O}$ bond; the peak at 703 cm^{-1} is attributable to the $\text{A}_{1g}(1)$ vibration mode of the Fe-O symmetric stretching. All these peaks are similar to those

of a CoFe_2O_4 spinel crystal found in the literature³⁴ and this confirms our interpretation of TGA and PRXD analysis. This analysis also confirms that the peaks at 202 cm^{-1} ; 263 cm^{-1} ; 300 cm^{-1} ; 318 cm^{-1} ; 358 cm^{-1} ; 610 cm^{-1} ... in agreement with the literature,³⁵ show the presence (in the trace state) of the Fe_5LiO_8 materials ascribed to the PXRD analysis. This analysis also confirms that the residues obtained from the thermal decomposition of the Co and Fe precursors are, respectively, Co_3O_4 (see Fig. S7, ESI†) and a mixture of Fe_2O_3 and Fe_3O_4 (see Fig. S8, ESI†) in agreement with the TGA and PXRD results.

3.3.3. Energy dispersive X-ray (EDX) analysis of CoFe_2O_4 . The chemical composition of the elements that constitute the crystalline phase of cobalt ferrite, obtained experimentally, is determined and presented in the spectral analysis shown in Fig. 5.

The EDX spectra clearly confirmed the presence of the expected elements (Fe, Co and O). Moreover, the semi-quantitative analysis performed indicates that the experimental Fe/Co ratio obtained is 1.8, which is close to the expected value (~ 2). The slight difference observed between the expected and obtained values can be explained by the presence of some impurities such as residual Cl and C. This result confirms the hypothesis that the synthesized material is CoFe_2O_4 .

3.3.4. SEM result of the CoFe_2O_4 . In order to determine the morphology and approximate particle size of the synthesized oxides, photographs were taken using a scanning electron microscope. Fig. 6 presents the SEM image of the cobalt ferrite and Fig. S9 and S10 (ESI†) show the SEM image of Co_3O_4 and $\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4$, respectively.

It can be observed that cobalt ferrite has a polyhedral-like shape with a size ranging from 100–300 nm, almost similar to that obtained for iron oxide. However, cobalt oxide exhibited mainly spherical particles with an average grain size of less than 100 nm.

3.4. Electrocatalytic HER performance of the synthesized CoFe_2O_4

Fig. 7 presents the HER characteristics of the experimentally synthesized material.

The catalytic performance of the synthesized material with respect to hydrogen release reactions is elucidated in Fig. 7. Fig. 7(a) displays the LSV plots of different CoFe_2O_4 electrodes and those of reference (GC). The CoFe_2O_4 electrocatalyst synthesized in this work required an overpotential of 440 mV to reach a current density of 10 mA cm^{-2} , a satisfactory HER

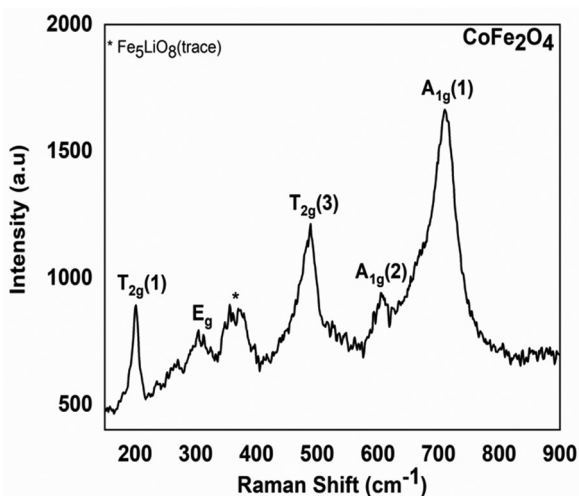


Fig. 4 Experimental Raman spectrum of the residue from the thermal decomposition of the Co-Fe precursor.

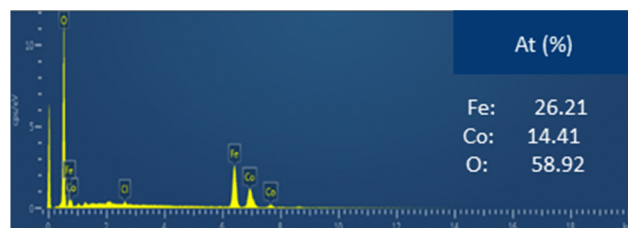


Fig. 5 Energy dispersive X-ray analysis of the decomposition residue.



Fig. 6 Left: SEM image of (left) CoFe_2O_4 on a reduced scale ~ 200 nm; Right: enlargement of the framed part of left image.

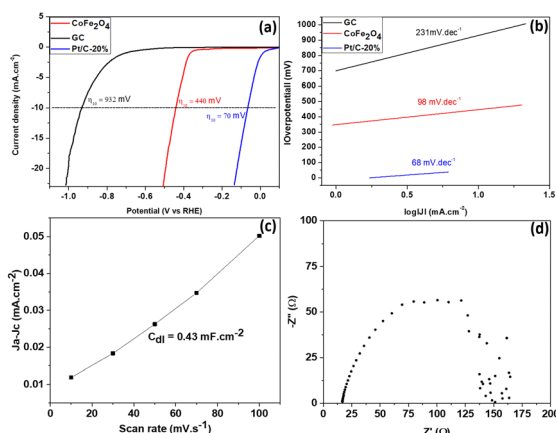
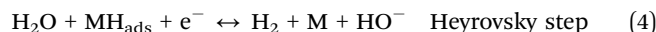
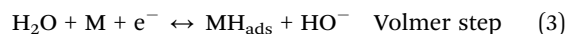


Fig. 7 (a) LSV curve of CoFe_2O_4 . (b) Tafel plot of CoFe_2O_4 . (c) C_{dl} plot of CoFe_2O_4 obtained from the cyclic voltammetric curve at different sweep speeds. (d) Nyquist plot of CoFe_2O_4 .

value if we stick to an application offered by a material resulting from green synthesis. This value is comparable and more or less lower than those of Pt/C-20% (70 mV), NiFe_2O_4 (420 mV) and ZnFe_2O_4 (520 mV) obtained by Hamza Belhadj *et al.* (using chemical reagents).¹ Similar work on nickel ferrite doped with cobalt at x% reports overpotentials, almost all of which are between 450 and 600 mV at a current density of 10 mA cm^{-2} .³⁶ This would justify the quality of our synthesized material compared to those which would be located in this zone of the overpotentials mentioned above. From the point of view of the kinetics limiting the dihydrogen evolution reaction process, the value of the Tafel slope (which is 98 mV dec^{-1} as depicted on Fig. 7(b)) is significantly lower than that of CuFe_2O_4 (124 mV dec^{-1}), NiFe_2O_4 (133 mV dec^{-1}) and ZnFe_2O_4 (144 mV dec^{-1}), obtained by Hamza Belhadj *et al.*¹ This shows that the adsorption of H_2O is energetically less favorable, according to the Volmer mechanism (eqn (3)), on the synthesized CoFe_2O_4 compared to the ferrites of CuFe_2O_4 , NiFe_2O_4 and ZnFe_2O_4 obtained by Hamza Belhadj *et al.*¹ Very recent work on the study of the electrocatalytic performance of NiCo_2O_4 nanostructures prepared by the hydrothermal method and post-annealing heat treatment, presented the Tafel of various samples of NiCo_2O_4 nanostructures annealed at different temperatures. It follows from this optimization that almost all of these samples have Tafel between 98 mV dec^{-1} and 168 mV dec^{-1} ,³⁷ which is higher compared to the value found experimentally in our work. The value of the Tafel slope obtained experimentally suggests that dihydrogen is produced

according to the Volmer–Heyrovsky mechanism.^{38,39} The first step, called the Volmer step, consists of the adsorption of a molecule of water on the surface of a catalytic site M. The second step is a step of recombination of the previously adsorbed hydrogen atom with a hydrogen atom resulting from the dissociative adsorption of a water molecule (Heyrovsky step). Eqn (3) and (4) illustrate the rate limiting step of the process (rate determining step), determined from the Tafel slope.



The dihydrogen produced from the electrocatalysis of water by the synthesized material offers a promising prospect for the green synthesis of CoFe_2O_4 as a cost-effective ecological electrocatalyst for an electrochemical reaction of hydrogen evolution. Fig. 7(c) illustrates the linear relationship between the current density difference and the scan rate at a potential of 0.05 V. This plot is obtained from the cyclic voltammetric curves at different scan rates (see Fig. S11 for additional information, ESI†). The C_{dl} value of 0.43 mF cm^{-2} obtained in this work is comparable to the 0.412 mF value for ZnFe_2O_4 obtained by Hamza *et al.*¹ and the C_{dl} values include between 0.0064 and 0.318 mF cm^{-2} for various samples of NiCo_2O_4 nanostructures annealed at different temperatures, obtained by Azhar *et al.*³⁷ This result reveals a non-negligible number of active sites exposed to reactions on the surface of the synthesized material. This confirms the electrochemical activity of the as prepared CoFe_2O_4 which is in agreement with previous results. The Nyquist plot (Fig. 7(d)) obtained from the EIS measurements revealed that the charge transfer resistance of the CoFe_2O_4 catalyst (147 Ω) is comparable to that of CuFe_2O_4 (120.7 Ω) and much smaller than that of NiFe_2O_4 (265.7 Ω) and ZnFe_2O_4 (486.1 Ω) obtained by Hamza *et al.*¹ This indicates a higher electronic conductivity, fastest charge transfer rate and easier electrocatalytic kinetics of CoFe_2O_4 synthesized with the aforementioned ferrites. Furthermore, although the characteristics of the HER activity of the material synthesized in this work are appreciable, they are not optimal when compared to that of the material synthesized from chemical reagents. This is the case for example of cobalt ferrite, obtained by coprecipitation, by Hamza Belhadj *et al.*, having an overpotential of 270 mV, a value of the Tafel slope of 98 mV dec^{-1} and a value of $C_{\text{dl}} = 2.127 \text{ mF cm}^{-2}$,¹ and that obtained by a hydrothermal method, by Atano Panda *et al.*, having an overpotential of 370 mV, a value of the Tafel slope of 283 mV dec^{-1} and a value of $C_{\text{dl}} = 1.400 \text{ mF cm}^{-2}$.⁴⁰ However, the recovery of waste and the production cost are the assets for this work.

Concerning the long-term stability during the HER of the as prepared electrocatalyst, chronopotentiometry was performed at a fixed current density of 10 mA cm^{-2} for 12 h.

The result presented in Fig. 8a shows a small drop of 13.07 mV in the overpotential after 12 h, indicating, therefore, its good stability during the HER process. After the stability test, EIS was carried out on the sample and the result is shown in Fig. 8b. The R_{ct} calculated from the graph has a value of



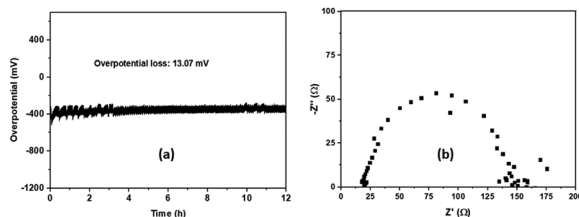


Fig. 8 (a) Chronopotentiometry of CoFe_2O_4 . (b) The electrochemical impedance spectrum, after the stability test.

146 U, which is almost similar to that calculated before the stability test (*cf.* 147 U), revealing the same charge transfer capacity and, therefore, no considerable alteration of the electrical properties. This demonstrates that the prepared material is an effective and stable electrocatalyst for the HER in alkaline medium.

4. Conclusion

In this research, the aim was to carry out the green synthesis of cobalt ferrite by a coprecipitation route by using rotten passion fruit juice. The compound with a prismatic shape and particle size of less than 300 nm, was successfully synthesized and characterized by XRD, Raman spectroscopy, EDX and SEM analyses. This route was also used to successfully synthesize the Co_3O_4 and a mixture of Fe_2O_3 and Fe_3O_4 . HER characteristics were assessed using LSV, Tafel slope (TS), electric double-layer capacitance (C_{dl}), and EIS. The CoFe_2O_4 electrocatalyst synthesized in this work required an overpotential of 440 mV; a TS of 98 mV dec^{-1} , a $C_{dl} = 0.43 \text{ mF cm}^{-2}$, a charge transfer resistance $R_{ct} = 147 \text{ U}$ and good stability with an overpotential loss of only 13.07 mV after 12 h. By comparing these data with those found in the literature, we can conclude that the CoFe_2O_4 electrocatalyst synthesized from the rotten passion juice fruit is a suitable catalyst for the hydrogen evolution reaction, which is a renewable and clean energy source for the environment. These characteristics could be improved by adjusting certain parameters such as the calcination temperature and the cooling time of the material after calcination. Such synthesis from the juice of rotten passion fruits is easy, sustainable and it can be an illustration of a promising step towards a totally green technology.

Author contributions

Rochelin Prosper Medang: investigation and draft preparation. Roussin Lontio Fomekong: investigation, resources and writing – original draft. Hypolite Mathias Tedjieukeng Kamta: methodology, data curation and formal analysis. Cédrik Ngnintendem Yonti: data curation, visualization and formal analysis. Edwin Akongnwi Nforna: data curation, resources and formal analysis. Patrice Kenfack Tsobnang: conceptualization, methodology and editing. John Ngolui Lambi: supervision, reviewing

and editing. Dieudonné Bitondo: supervision, reviewing and editing.

Conflicts of interest

The authors declare that they have no known competing interests.

Acknowledgements

The authors acknowledge Professor Zdenek Sofer for the characterization facilities used in this work. The authors also acknowledge the RSC Research Fund Grant no. R23-1928872283 for electrochemical facilities.

References

- 1 B. Hamza, Y. Messaoudi, R. M. Khelladi and A. Azizi, *Int. J. Hydrogen Energy*, 2022, **47**, 20129.
- 2 R. Bao-Ping, G. Yang, L. Zhou-Yan, L. Zheng-Yan, H. Zhang, S. Li-Ping and L. Hai-Yang, *Int. J. Hydrogen Energy*, 2023, **152**, 110663.
- 3 M. Nemiwal, V. Gosu, T. C. Zhang and D. Kumar, *Int. J. Hydrogen Energy*, 2021, **46**, 10216.
- 4 F. Dawood, M. Anda and G. M. Shafiullah, *Int. J. Hydrogen Energy*, 2020, **45**, 3847.
- 5 C. Li and J. B. Baek, *ACS Omega*, 2020, **5**, 31–40.
- 6 J. Li, S. Wang, J. Chang and L. Feng, *Adv. Powder Mater.*, 2022, **1**, 00030.
- 7 D. Li, H. Liu and L. Feng, *Energy Fuels*, 2020, **34**, 13491.
- 8 D. Y. Li, L. L. Liao, H. Q. Zhou, Y. Zhao, F. M. Cai, J. S. Zeng, F. Liu, H. Wu, D. S. Tang and F. Yu, *Mater. Today Phys.*, 2021, **16**, 100314.
- 9 J. Yu, T. Le, N. Tran and H. Lee, *Chem. – Eur. J.*, 2020, **26**, 6423.
- 10 P. Babar, A. Lokhande, V. Karade, B. Pawar, M. G. Gang, S. Pawar and J. H. Kim, *Chem. Eng.*, 2019, **7**, 10035.
- 11 P. Babar, K. Patil, J. Mahmood, S. Kim, J. Kim and C. Yavuz, *Cell Rep. Phys. Sci.*, 2022, **3**, 100762.
- 12 P. Palade, C. Comanescu, A. Kuncser, D. Berger, C. Matei, N. Lacob and V. Kuncser, *J. Nanomater.*, 2020, **10**, 476.
- 13 E. Gopalan, P. Joy, I. Al-omari, D. Kumar, Y. Yoshida and M. Anantharaman, *J. Alloys Compd.*, 2009, **485**, 711–717.
- 14 O. Cofe and S. Jovanovic, *J. Alloys Compd.*, 2014, **589**, 271–277.
- 15 R. Sani, A. Beitollahi, Y. Maksimov and I. Suzdalev, *J. Mater. Sci.*, 2007, **42**, 2126–2131.
- 16 T. Prabhakaran and J. Hemalatha, *J. Alloys Compd.*, 2011, **509**, 7071–7077.
- 17 I. Gul, A. Maqsood, M. Naeem and M. Ashiq, *J. Alloys Compd.*, 2010, **507**, 201–206.
- 18 H. M. Tedjieukeng, P. T. Kenfack, R. L. Fomekong, E. P. Ekane, A. Pattayil, A. Delcorte and J. N. Lambi, *RSC Adv.*, 2018, **67**, 38621.



- 19 C. N. Yonti, P. K. Tsobnang, R. L. Fomekong, F. Devred, E. Mignolet, Y. Larondelle, S. Hermans, A. Delcorte and J. N. Lambi, *J. Nanomater.*, 2021, **11**, 2833.
- 20 E. P. Ekane, T. J. Foba, N. V. Beckley, Y. D. Mbom, H. M. Tedjieukeng, R. L. Fomekong and J. N. Lambi, *J. Nanomater.*, 2020, **7**, 8830136.
- 21 F. E. El-Belely, M. S. Farag, A. H. Said, A. S. Amin, E. Azab, A. A. Gobouri and A. Fouda, *J. Nanomater.*, 2021, **11**, 95.
- 22 S. A. Fahmy, I. M. Fawzy, B. M. Saleh, M. Y. Issa, U. Bakowsky and H. M. El-Said Azzazy, *J. Nanomater.*, 2021, **11**, 965.
- 23 A. Franco, R. Luque and C. Carrillo-Carrión, *J. Nanomater.*, 2021, **11**, 1259.
- 24 J. L. Coffey and L. T. Canham, *J. Nanomater.*, 2021, **11**, 553.
- 25 S. M. Alsaggaf, A. M. Diab, E. F. B. ElSaied, A. A. Tayel and H. S. Moussa, *J. Nanomater.*, 2021, **11**, 385.
- 26 F. Renia, V. Alexandra, K. Chatzi, H. Mohamed, N. Chalmes, D. Moschovas, K. Spyrou, C. Angeliki, A. Apostolos, D. Gournisand and H. Stamatis, *J. Nanomater.*, 2021, **11**, 458.
- 27 J. Favier, Irue de NRA-147, l'Université-75338 Paris Cedex 07, 1993.
- 28 J. Coates, Interprétation des spectres infrarouges, une approche pratique, in *Encyclopedia of Analytical Chemistry*, ed. R. A. Meyers, John Wiley & Sons Ltd, Chichester, 2000, pp. 10815–10837.
- 29 S. Dragan and A. Fitch, *J. Chem.*, 1998, **75**, 1018–1021.
- 30 M. A. Berube, D. Schorr, R. J. Ball, V. Landry and P. Blanchet, *J. Polym. Environ.*, 2018, **26**, 970–979.
- 31 P. M. Eduardo, S. D. Letícia, M. L. de Souza, J. K. Ribeiro, P. W. Marques, D. P. Rodrigo, S. S. Porto, C. J. R. Proveti and C. E. Passamani, *J. Cleaner Prod.*, 2020, **265**, 121712.
- 32 S. S. Banifatemi, F. Davar, B. Aghabarari, A. J. Segura, J. F. Alonso and M. S. Ghoreishi, *Ceram. Int.*, 2021, **47**, 19198–19204.
- 33 K. Kombaiah, V. J. Judith, L. J. Kennedy, M. Bououdina, R. J. Ramalingam and H. Al-Lohedan, *Mater. Chem. Phys.*, 2017, **204**, 410–419.
- 34 P. Chandramohan, M. Srinivasan, S. Velmurugan and S. Narasimhan, *J. Solid State Chem.*, 2011, **184**, 89–96.
- 35 W. Cook and M. Manley, *J. Solid State Chem.*, 2010, **183**, 322–326.
- 36 C. Sanaz, K. Maasoumeh, S. P. Naeimeh and A. Umut, *ACS Omega*, 2021, **6**, 33024–33032.
- 37 S. Azhar, S. Sumaiya, U. Noor, S. Muhammad, L. Yihan, K. Amir and K. Majid, *J. Mater. Sci. Eng. B*, 2023, **294**, 116508.
- 38 A. Lasia, in *Handbook of Fuel Cells*, ed. W. Vielstich, A. Lamm, H. A. Gasteiger, and H. Yokokawa, 2003, pp. 414–440.
- 39 G. Q. Han, X. Li, Y. R. Liu, B. Dong, W. H. Hu, X. Shang, X. Zhao, Y. M. Chai, Y. Q. Liu and U. C. G. Liu, *RSC Adv.*, 2016, **6**, 52761.
- 40 A. Panda, H. Kyu and H. Kim, *Int. J. Mol. Sci.*, 2023, **24**, 9585.

