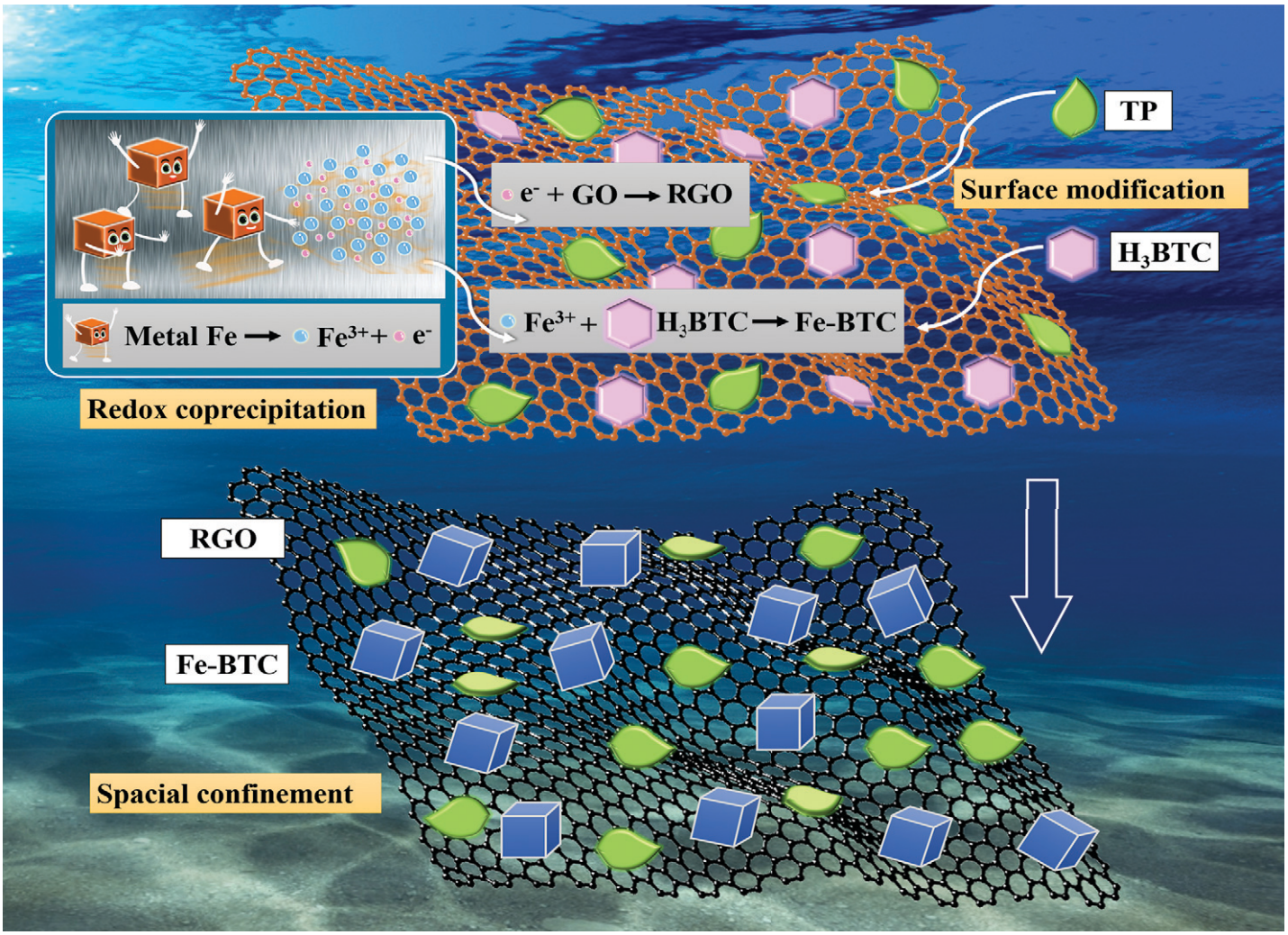


Showcasing research from Professor Binghui Xu's laboratory, Institute of Materials for Energy and Environment, College of Materials Science and Engineering, Qingdao University, Qingdao 266071, China.

Synthesis of a MOF-derived magnetite quantum dots on surface modulated reduced graphene oxide composite for high-rate lithium-ion storage

Reactions between 1,3,5-benzenetricarboxylic acid (H_3BTC), tea polyphenol (TP), graphene oxide (GO) and iron foils are triggered under mild aqueous conditions. Both H_3BTC and TP are decorated on the surface of GO sheets by π–π conjugation while the redox reaction between Fe and GO takes place. The coordination between Fe^{3+} and BTC^{3+} leads to *in situ* crystallization of Fe-BTC metal–organic framework (MOF) on the surface of the TP-decorated reduced graphene oxide (TP-RGO), both the size of Fe-BTC MOF domains and the restacking of TP-RGO sheets are significantly controlled.

As featured in:

See Binghui Xu *et al.*, *RSC Appl. Interfaces*, 2024, **1**, 233.

ROYAL SOCIETY OF CHEMISTRY

rsc.li/RSCApplInter

Registered charity number: 207890

RSC Applied Interfaces

ROYAL SOCIETY OF CHEMISTRY

PAPER

View Article Online
View Journal | View Issue

Check for updates

Cite this: *RSC Appl. Interfaces*, 2024, 1, 233

Synthesis of a MOF-derived magnetite quantum dots on surface modulated reduced graphene oxide composite for high-rate lithium-ion storage

Ruixin Jia, Longbiao Yu, Zhenqi Han, Shuo Liu, Panpan Shang, Siqi Deng, Xuehua Liu and Binghui Xu *

Received 6th August 2023,
Accepted 3rd October 2023

DOI: 10.1039/d3lf00128h

rsc.li/RSCApplInter

As an anodic candidate material for lithium-ion batteries (LIBs), magnetite (Fe_3O_4) has the advantages of high theoretical capacity (926 mA h g^{-1}), environmental friendliness, and natural abundance. The critical problems of inferior rate capability and cycling stability for Fe_3O_4 anodes need to be urgently solved. In this work, a composite anode material with Fe_3O_4 quantum dots immobilized by pyrolytic carbon and reduced graphene oxide skeleton (Fe_3O_4 QDs@C/RGO) is rationally engineered from a metal–organic framework (MOF) domain on surface-modulated RGO precursor. The involved raw materials are 1,3,5-benzenetricarboxylic acid ($C_9H_6O_6$, trimesic acid, H_3BTC), tea polyphenol (TP), few-layered graphene oxide (GO) and metal iron (Fe) foils, which react in mild hydrothermal condition in the synthesis of the Fe-BTC/TP-RGO precursor sample. The essence of the reactions has been uncovered, and the Fe_3O_4 QDs@C/RGO composite shows excellent lithium-ion storage capability with a reversible capacity of 1070.84 mA h g^{-1} after 150 cycles at 200 mA g^{-1}, 738.03 mA h g^{-1} after 300 cycles at 1000 mA g^{-1} in half cells, and 717.29 mA h g^{-1} after 150 cycles at 200 mA g^{-1} in full cells. The sample synthesis protocol has the merits of simplicity and eco-friendliness, and the Fe_3O_4 QDs@C/RGO sample can be regarded as a promising anodic candidate for high-performance LIBs.

1. Introduction

Lithium-ion batteries (LIBs) currently have a dominant position in the portable electronics and new-energy automobile markets primarily from their higher energy density and superior cycling performance.[1–3] To overcome the obstacle of the theoretical capacity limit of graphitic anodes (372 mA h g^{-1}), both academia and industry are striving to develop advanced anode materials to satisfy the increasing demand from the rapid development of renewable energy utilization.[4–12] Magnetite (Fe_3O_4) has been extensively studied as an alternative anode material for LIBs due to its high theoretical capacity (926 mA h g^{-1}), environmental friendliness, and natural abundance. However, the sluggish electron conductivity and heavy volume change always lead to inferior cycling stability, particularly under higher current rate.[13,14] In recent studies, quantum dots (QDs) with ultra-small size as the electrochemically active materials in LIBs have been proven to be able to shorten the ion diffusion distance, reduce the internal strain, and improve the reaction conversion efficiency.[15–18] Meanwhile, to achieve satisfactory lithium-ion storage performances at high current densities, effectively immobilizing Fe_3O_4 QDs with a unique conductive supporting framework is a challenging task.

Two-dimensional graphene oxide (GO) with a large specific surface area and abundant functional groups is dispersible in water phase, which facilitates the synthesis of reduced graphene oxide (RGO)-supported metal oxide nanocomposite materials.[19,20] However, the excessive size growth of the nanocrystals and over-restacking of the RGO sheets impedes to a large extent the engineering of Fe_3O_4 QDs on RGO with optimized microstructure. On the other hand, porous metal–organic framework (MOF) materials are usually constituted of periodically arranged metal ion centers and organic ligands from a coordinating chelation reaction, which are also being employed as precursors for the synthesis of pyrolytic carbon-supported metal oxide nanocomposites.[21–25] In our previous work, 1,3,5-benzenetricarboxylic acid ($C_9H_6O_6$, H_3BTC), a water-soluble small-molecule organic ligand, has been introduced in a redox coprecipitation reaction between metallic Fe and GO sheets to synthesize Fe-BTC MOF domains on a RGO sheets precursor sample, which is finally converted to Fe_3O_4 nanoparticles on carbon matrix after thermal treatment.[26] Therefore, it is of great significance to further explore a simplified and eco-friendly method to synthesize a

Institute of Materials for Energy and Environment, College of Materials Science and Engineering, Qingdao University, Qingdao 266071, China.
E-mail: xubinghuiqdu@qdu.edu.cn, xubinghuicsu@163.com

carbonaceous matrix-supported Fe_3O_4 QDs composite on the above foundation.

Tea polyphenol (TP) is an extract from green tea and a water-soluble mixture of condensed tannins with main-chain molecules consisting mainly of rigid aromatic rings. TP is effective in modifying the surface properties of RGO sheets, which can act as a reducing agent for partial removal of the functional groups on GO and directly decorate on RGO sheets.[27–29] In this work, as illustrated in Scheme 1, under mild aqueous conditions, H_3BTC, TP, GO, and iron foils are directly used as starting materials. Both H_3BTC and TP molecules are decorated on the surface of the GO sheets *via* the π–π conjugation effect while a redox reaction between the Fe atoms and the oxidative functional groups of GO sheets takes place. Moreover, a chelating reaction between Fe^{3+} and the BTC^{3+} leads to the *in situ* crystallization of the Fe-BTC MOF domains on the TP-decorated RGO (TP-RGO) framework. Particularly, in the presence of TP, both the crystallization of the Fe-BTC MOF domains and the restacking of the TP-RGO sheets are significantly controlled. In other words, the as-formed small-sized Fe-BTC domains and abundant TP molecules are jointly dispersed on RGO sheets without severe restacking in the Fe-BTC/TP-RGO precursor. After thermal treatment, the composite sample with Fe_3O_4 QDs immobilized by pyrolytic carbon and RGO skeleton (Fe_3O_4 QDs@C/RGO) is finally synthesized, in which the homogeneously distributed small-sized Fe-BTC domains are converted to pyrolytic carbon-immobilized Fe_3O_4 QD clusters on RGO sheets. The unique microstructure of the Fe_3O_4 QDs@C/RGO sample contributes to improved electronic and ionic transportation efficiency and suppressed volume change; thus, significantly enhanced electrochemical performances for this sample have been witnessed.

2. Experimental section

2.1 Reagents and materials

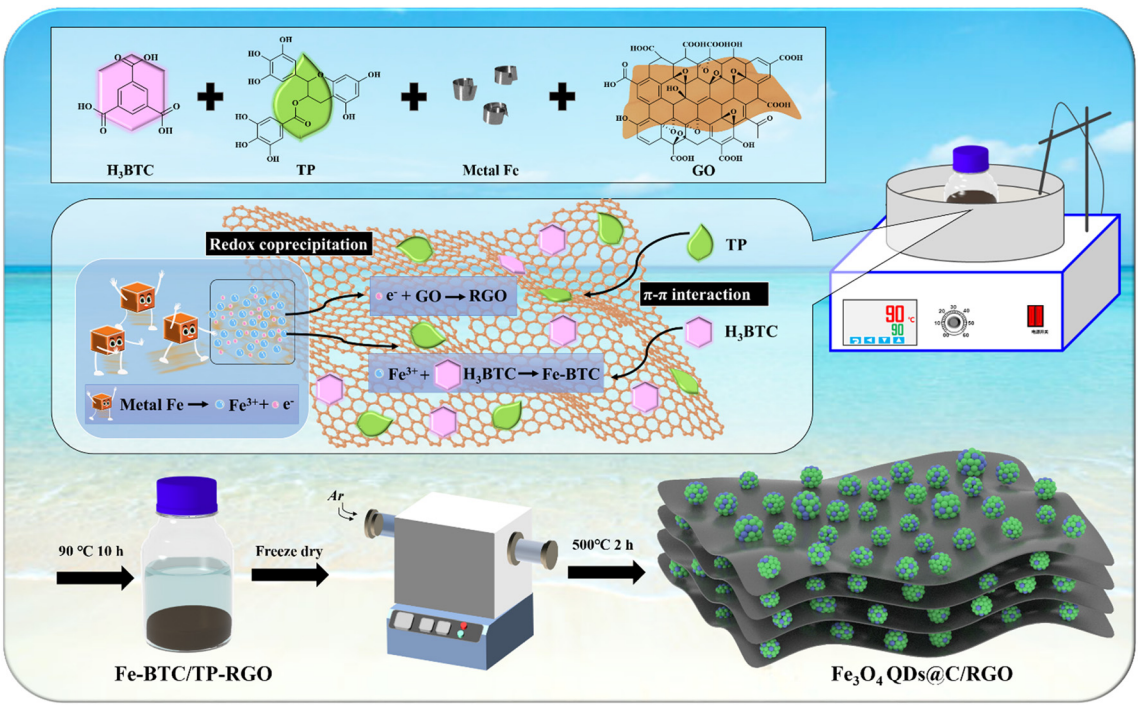
H_3BTC (98%) was supplied by Accela Chembio. TP (97%) was obtained from Macklin. Metal Fe foil (0.15 mm thick, 99.99%) was purchased from Hunan Metal Materials, China. GO was synthesized from natural graphite flake (Sigma Aldrich, 325 mesh) according to our previously published method.[30]

2.2 Sample synthesis

Fe_3O_4 QDs@C/RGO: in a typical experiment, H_3BTC (1.0 g), TP (1.0 g), GO (0.1 g) and several polished iron foils were dissolved in 150 mL of deionized water and kept magnetically stirred in an oil bath at 90 °C for 10 h. After freeze-drying, the intermediate Fe-BTC/TP-RGO powder with black color was obtained. Finally, the Fe-BTC/TP-RGO sample underwent a thermal treatment at 500 °C for 2 h in Ar condition, and the final black Fe_3O_4 QDs@C/RGO sample was thus fabricated.

2.3 Sample characterization and electrochemical measurement

2.3.1 Sample characterization. X-ray diffraction (XRD) patterns were measured with an X-ray diffractometer (Rigaku Ultima IV with Cu Kα radiation, λ = 0.15418 nm). Fourier transform infrared spectroscopy (FTIR) was conducted in the



Scheme 1 Schematic illustration for the synthesis of the Fe_3O_4 QDs@C/RGO sample.

region 400–4000 cm^{-1} using a Nicolet iS50 FTIR spectrophotometer. A Renishaw inVia Plus Micro-Raman spectroscopy system equipped with a 50 mW DPSS laser at 532 nm was used to record the Raman spectra of the samples. Thermogravimetric analyses (TGA) were performed with a Mettler Toledo TGA-2 in air with a heating rate of 10 °C min^{-1}. A PHI Quantera II spectrometer was used to carry out X-ray photoelectron spectroscopy (XPS). A Quantachrome Autosorb-IQ3 was used to measure the nitrogen adsorption/desorption isotherms, and the specific surface area was based on the Brunauer–Emmett–Teller (BET) method. Moreover, the corresponding pore size distributions were calculated based on the Barrett–Joyner–Halenda (BJH) model. Field-emission scanning electron microscopy (FESEM, JEOL Model JSM-7800F) and transmission electron microscopy (TEM, JEOL Model JEM-2100) were used to reveal the morphologies and microstructures.

2.3.2 Electrochemical measurement. For the assembly of coin-type half cells, a slurry was mixed with Fe_3O_4 QDs@C/RGO electrode materials, Super P, polyvinylidene fluoride (PVDF), and carbon fiber (CF) in a weight ratio of 7 : 1 : 1 : 1. An appropriate amount of 1-methyl-2-pyrrolidone (NMP) was added to form a homogeneous slurry, which was coated on copper foil by the doctor blade technique. For the $LiFePO_4$ cathode, $LiFePO_4$, Super P and PVDF were mixed in a weight ratio of 8 : 1 : 1, then NMP was added to form a homogeneous slurry, which was coated on an aluminum foil. CR2016 half cells were assembled with metal lithium foil as counter/reference electrode. Celgard 2600 film was used as a separator between the two electrodes. The electrolyte was composed of 1.0 M lithium hexafluorophosphate ($LiPF_6$) in a solvent of ethylene carbonate (EC), dimethyl carbonate (DMC), and ethylene methyl carbonate (EMC) in a volumetric ratio of 1 : 1 : 1, in which 1.0% vinylene carbonate (VC) was added. For the coin-type full cells, CR2025 cells were assembled using a $LiFePO_4$ cathode and an Fe_3O_4 QDs@C/RGO anode, in which the anode was pre-lithiated for 6 cycles in a half cell.

Cyclic voltammetry (CV) measurements were performed with a CHI 660E electrochemical workstation. Galvanostatic charge and discharge tests were carried out with a Neware battery test instrument and the specific capacity was calculated based on the mass of the synthesized composite samples. Electrochemical impedance spectroscopy (EIS) tests were also carried out using the CHI 660E electrochemical workstation.

3. Results and discussion

3.1 Structure and morphology

In the first mild aqueous condition, the color of the reaction system gradually turns black, and the Fe-BTC/TP-RGO sample can be collected as a sediment. After freeze-drying and thermal treatment, the final Fe_3O_4 QDs@C/RGO sample can be synthesized from the Fe-BTC/TP-RGO precursor. From the XRD pattern of the Fe-BTC sample in Fig. 1, several weak diffraction peaks located at 2θ = 10.16, 10.90, 20.06, and 23.98 degrees are in general agreement with Fe-BTC MOF.[31–33] The XRD pattern of Fe-BTC/TP-RGO displays a broad diffraction peak at 2θ = 24.50 degrees indicating the formation of the RGO supporting skeleton.[34] Moreover, from the locally magnified XRD patterns for the above two samples, the characterization peaks are in good agreement, indicating the successful engineering of the Fe-BTC MOF domains in the Fe-BTC/TP-RGO sample. However, the significantly weakened intensity of the XRD peaks for the Fe-BTC MOF species is highly attributed to its limited crystallization on the surface of the TP-modulated RGO sheets. For the Fe_3O_4 QDs@C/RGO sample, the apparent diffraction peaks situated at 2θ of 30.10, 35.42, 43.05, 53.39, 56.94, and 62.52 degrees correspond to the Fe_3O_4 (JCPDS card number 19-0629) crystallographic planes of (2 2 0), (3 1 1),

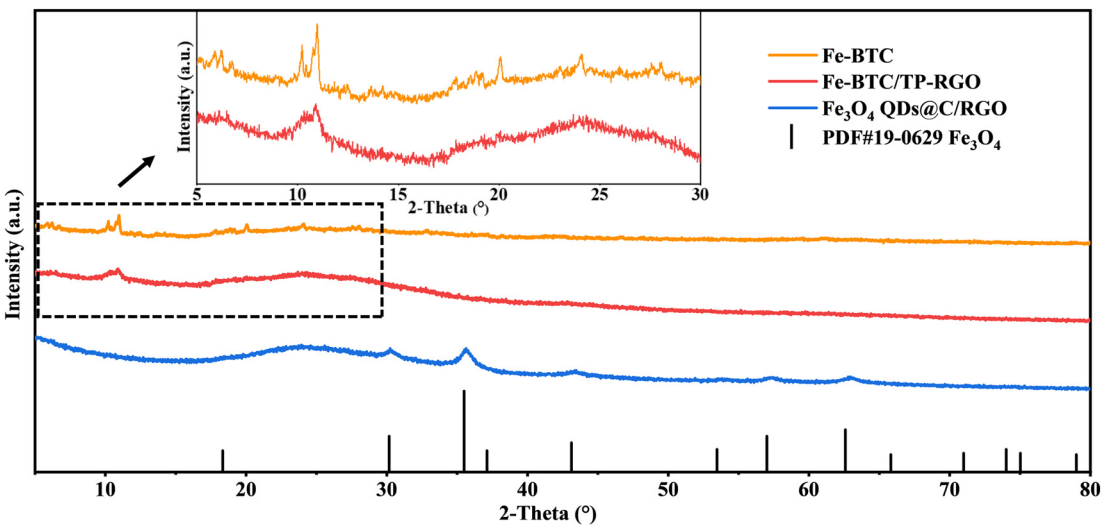


Fig. 1 XRD patterns of the Fe-BTC, Fe-BTC/TP-RGO and Fe_3O_4 QDs@C/RGO samples.

(4 0 0), (4 2 2), (5 1 1), and (4 4 0), respectively. The characterization peak of RGO can also be seen without any diffraction peaks of other possible impurities.

The FTIR analysis result of the Fe-BTC/TP-RGO sample is shown in Fig. 2a. The sharp peak at about 3388 cm^{-1} is attributed to free O–H bonds in the structure. The bands at 2852 cm^{-1}, 1097 cm^{-1} and 1636 cm^{-1} are attributed to the stretching vibrational peaks of C–H, C–O–C and C=C, while the peaks at 1636 and 1386 cm^{-1} belong to aromatic C=C bonds and C–OH bonds, respectively.[35,36] Moreover, the characteristic peak at 532 cm^{-1} is related to Fe–O stretching vibrations.

In Fig. 2b, two main Raman peaks of the D-band (1325 cm^{-1}) and G-band (1590 cm^{-1}) can be seen for the Fe-BTC/TP-RGO and Fe_3O_4 QDs@C/RGO samples. The D-band corresponds to the defective and disordered part of the sp^3 carbon, while the G-band is associated with the ordered sp^2 carbon.[37,38] The integral intensity ratio of the D-band and G-band is another feature of the Raman spectrum, which provides useful information about the degree of carbon disorder. The intensity ratios I_D and I_G are measured to be 0.94 and 0.64 for the two samples, indicating fewer defects and a high degree of carbon order after thermal treatment. It can be revealed that with the presence of the surface-decorated TP, the conjugated electronic structure of the RGO sheets is repaired during the thermal treatment. Moreover, the large amount of ligand BTC pyrolyzed carbon also accounts for the high graphitization degree of the Fe_3O_4 QDs@C/RGO sample.

Fig. 2c exhibits the TGA and DTG curves of the Fe_3O_4 QDs@C/RGO sample. From the TGA curve, the initial weight loss that occurred from 30 °C to 100 °C is ascribed to the evaporation of water. The subsequent weight loss mainly occurs between 300 °C and 600 °C, which can be attributed to the oxidation of Fe_3O_4 to Fe_2O_3 and the decomposition of pyrolytic carbon. The mass loss of the Fe_3O_4 QDs@C/RGO sample can be divided into two stages, probably originating from the decomposition of the organic pyrolyzed carbon and RGO sheets. From the corresponding DTG curve, the inflection points of the two stages are measured to be 369.2

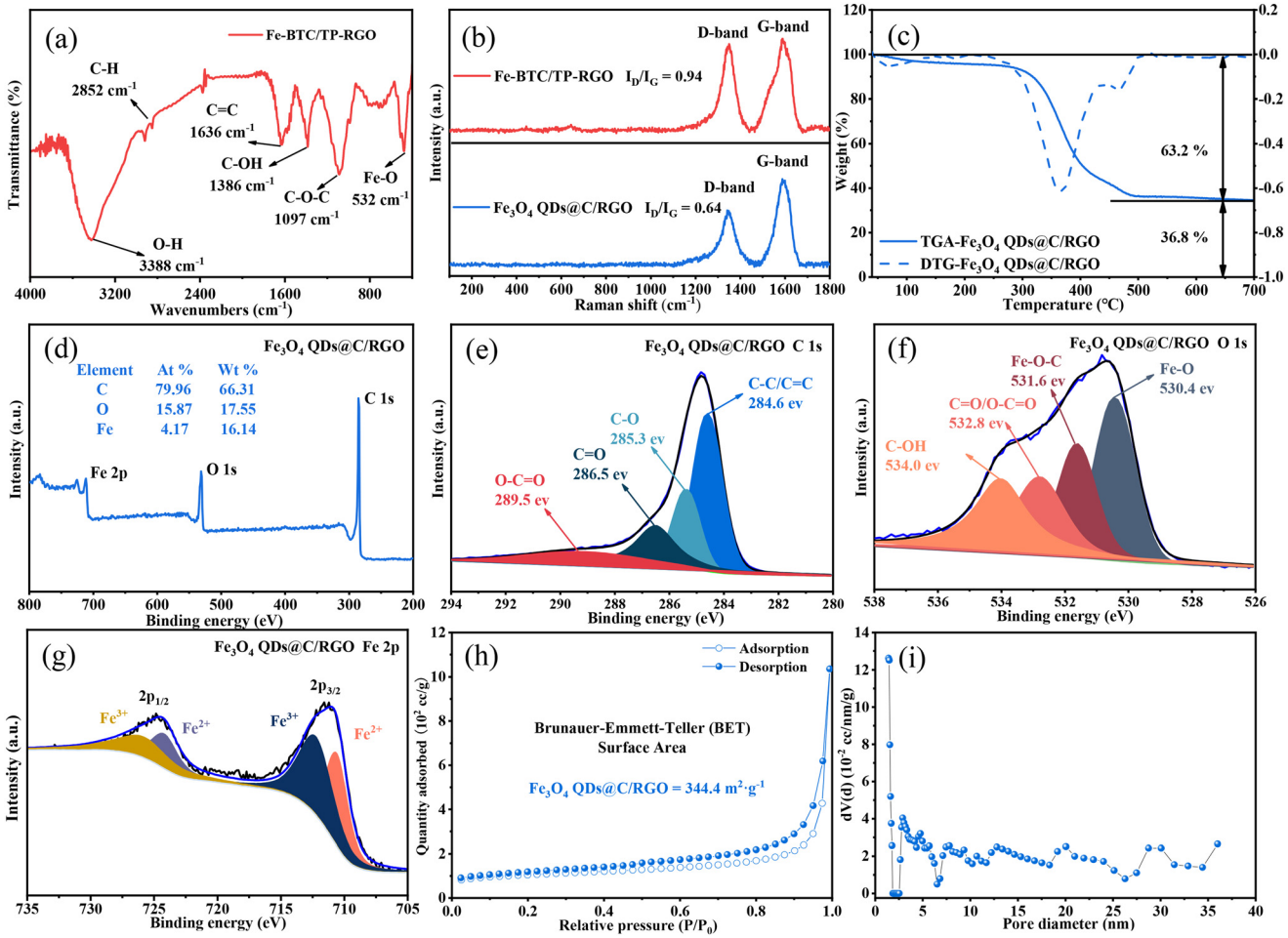


Fig. 2 FTIR pattern (a) of the Fe-BTC/TP-RGO sample. Raman spectra (b) of the Fe-BTC/TP-RGO and Fe_3O_4 QDs@C/RGO samples. TGA and DTG curves (c) of the Fe_3O_4 QDs@C/RGO sample. Survey XPS spectrum with atomic percentage summary (d). High-resolution C 1s (e), O 1s (f), Fe 2p (g) XPS spectra for the Fe_3O_4 QDs@C/RGO sample. N_2 adsorption and desorption isotherms (h) and pore size distribution (i) of the Fe_3O_4 QDs@C/RGO sample.

°C and 469.2 °C, respectively. The Fe_2O_3 remnant is stabilized at 36.8%. Therefore, the corresponding Fe_3O_4 QDs can be calculated to be 35.6% by mass while the carbon content is 64.4% in the Fe_3O_4 QDs@C/RGO composite.

The chemical valence and composition can be determined using the XPS technique. Fig. 2d shows the survey spectrum of the Fe_3O_4 QDs@C/RGO composite, which consists of the three elements C, O, and Fe with corresponding weight percentages of 66.31%, 17.55%, and 16.14%, respectively. This result reveals the high purity of the Fe_3O_4 QDs@C/RGO composite. As shown in Fig. 2e, the high-resolution C 1s spectrum can be resolved into four fitted peaks at 284.6, 285.3, 286.5, and 289.5 eV, which are attributed to C–C/C═C, C–O, C═O, and O–C═O bonds, respectively.[39,40] From the magnified O 1s spectrum in Fig. 2f, the peaks at 530.4, 531.5, and 533.3 eV are ascribed to Fe–O, Fe–O–C, and C–O/C═O bonds, respectively. The strong interaction between Fe_3O_4 and carbon matrix *via* the Fe–O–C linkage will be beneficial to maintain the structural stability of Fe_3O_4 QDs@C/RGO in the processes of discharging and charging as an anode material for LIBs.[41,42] Fig. 2g exhibits the high-resolution Fe 2p spectrum with two peaks located at 724.6 and 711.0 eV. For the fitting curves, two dominant peaks located at 726.0 and 712.4 eV are ascribed to the Fe $2p_{1/2}$ and Fe $2p_{3/2}$ states of Fe(III), while the other two peaks at 724.4 and 710.8 eV correspond to the Fe $2p_{1/2}$ and Fe $2p_{3/2}$ states of Fe(II), respectively.

To obtain the pore size distribution information and specific surface area of Fe_3O_4 QDs@C/RGO samples, nitrogen adsorption and desorption isotherms have been recorded. As shown in Fig. 2h, the isotherms are of type IV with a hysteresis loop between 0.4 and 1.0 relative pressure, indicating the mesoporous nature of the sample.[43,44] In addition, the BET specific surface area of the Fe_3O_4 QDs@C/RGO sample is estimated to be 344 $m^2\ g^{-1}$. The larger specific surface area facilitates the contact between electrolyte and active material.[45,46] As seen from BJH pore-size distribution curves in Fig. 2i, the overwhelming existence of mesopores with a wide range of pore diameters for the Fe_3O_4 QDs@C/RGO composite is verified.

FESEM images at different magnifications are used to investigate the morphological changes from the Fe-BTC/TP-RGO precursor to the final Fe_3O_4 QDs@C/RGO sample. From the FESEM images of the Fe-BTC/TP-RGO precursor in Fig. 3a–c, continuous large-sized RGO framework with corrugated microstructures can be clearly seen. Moreover, no obvious restacking and the smooth surface of the RGO sheets can also be verified. These results confirm the significantly controlled crystallization of the Fe-BTC domains on the surface of the RGO sheets, which is favorable for the engineering of the Fe_3O_4 QDs@C/RGO composite. After thermal treatment, the thin and wrinkled RGO framework can also be seen in Fig. 3d. In Fig. 3e and f, well-distributed small clusters are uniformly dispersed and anchored on the

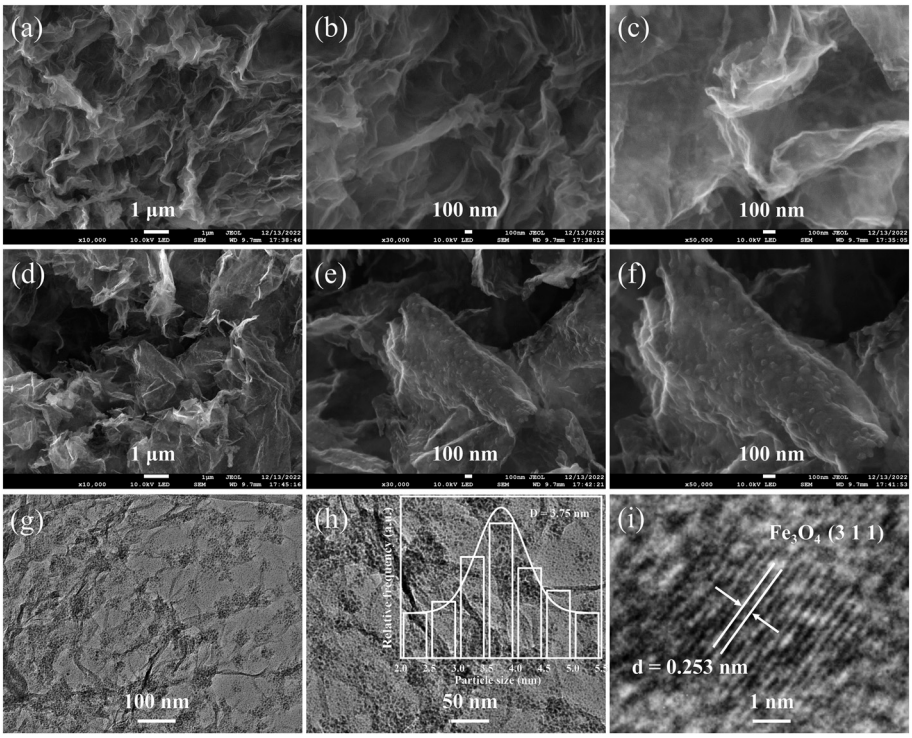


Fig. 3 FESEM images of Fe-BTC/TP-RGO (a–c) and Fe_3O_4 QDs@C/RGO (d–f) samples under various magnifications. TEM images (g and h) of the Fe_3O_4 QDs@C/RGO sample under different magnifications with inset showing Fe_3O_4 particle size distribution. HRTEM image (i) for an individual Fe_3O_4 nanoparticle in the Fe_3O_4 QDs@C/RGO sample.

RGO layer, which is highly related to the pyrolysis of the decorated Fe-BTC domains and induced size growth for the corresponding Fe_3O_4@C domains. From the TEM images of the Fe_3O_4 QDs@C/RGO sample in Fig. 3g and h, small Fe_3O_4 QDs locally assembled in a cluster configuration can be observed on the surface of the wrinkled RGO sheets. Most of the Fe_3O_4 QD-containing clusters are below 100 nm, and obvious space between the Fe_3O_4 QDs can also be verified. The particle size of Fe_3O_4 is accurately measured and the data nonlinearly fitted by the Gaussian algorithm, as shown in the inset image of Fig. 3h. The particle sizes are in the range of 2.0–5.5 nm, and the corresponding average particle size is calculated to be 3.75 nm. From the HRTEM image in Fig. 3i, the crystallographic spacing of an individual Fe_3O_4 QD is measured to be 0.253 nm, matching the (3 1 1) crystal plane of the Fe_3O_4 species.

3.2 Electrochemical characterization

Fig. 4a exhibits the CV profiles of the Fe_3O_4 QDs@C/RGO composite electrode for an initial five cycles with a scan rate of 0.1 mV s^{-1} in the potential range of 0.01 to 3.0 V (*vs.* Li/Li^+). In the curve of the initial cathodic scanning, a sharp peak at about 0.64 V can be observed, which corresponds to the reduction of Fe_3O_4 nanocrystals to metal Fe phase, accompanied by the generation of irreversible solid electrolyte interface (SEI) and the side reactions of electrolyte decomposition.[47,48] The obvious peak at about 0.01 V can be ascribed to the Li^+ insertion into the carbon matrix.[49,50] In the subsequent curves, the cathodic peak shifts slightly to a higher voltage of about 0.81 V, which can be ascribed to the enhanced polarization and structural modification of electrode material caused by Li^+ insertion-extraction during the initial cycle.[51,52] Moreover, the two anodic peaks at 1.59 and 1.88 V may originate from the oxidation of the metallic Fe phase.[53] It is noteworthy that the subsequent CV curves almost overlap, demonstrating the good reversibility of the Fe_3O_4 QDs@C/RGO composite electrode.

Fig. 4b shows the profiles of the constant current charge and discharge tests of the Fe_3O_4 QDs@C/RGO composite at 200 mA g^{-1}. The initial cycle provides charge and discharge capacities of about 1266.62 and 2098.59 mA h g^{-1}, respectively. Thus, the initial coulombic efficiency can be calculated to be 60.36%. The reason for the low initial coulombic efficiency is most likely caused by the formation of SEI.[54,55] The voltage plateaus for this electrode are in good agreement with the redox peaks in the CV curves. Fig. 4c shows the cycling performance of the Fe_3O_4 QDs@C/RGO electrode at a low-density current of 200 mA g^{-1}. The specific capacity is maintained at 1070.84 mA h g^{-1} after 150 cycles. Fig. 4d shows the long-time lithium-ion storage capability of the Fe_3O_4 QDs@C/RGO electrode at a high-density current of 1000 mA g^{-1}. At the 300th cycle, this electrode still provides a high reversible capacity of about 738.03 mA h g^{-1}. These results indicate the good stability of the Fe_3O_4 QDs@C/RGO electrode. Due to the formation of metallic iron, this electrode exhibits a gradual capacity fading in the first few cycles. At low potentials, the *in situ* formed iron atoms promote the generation of a conducting polymer film, which contributes to the increase of lithium-ion storage *via* a pseudocapacitive behavior. Accompanying the improved reaction conditions, metallic iron atoms gradually participate in reversible electrochemical reactions.[56,57] The above factors may

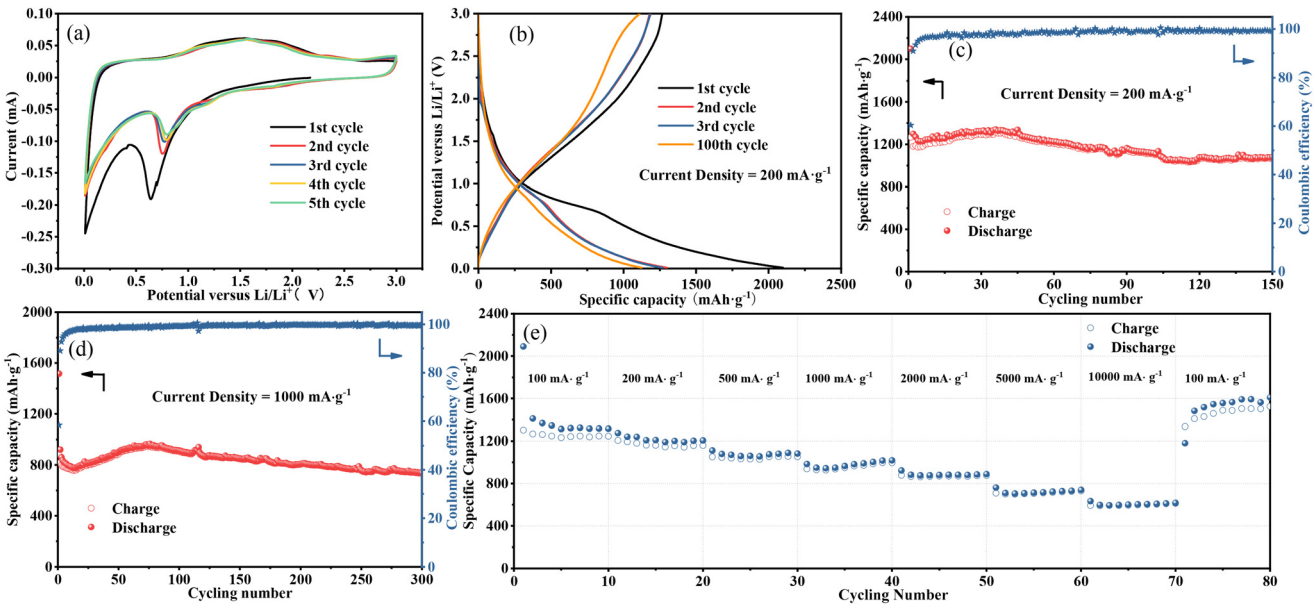


Fig. 4 The CV curves for the initial 5 cycles (a), and charge and discharge voltage profile (b) of the Fe_3O_4 QDs@C/RGO composite. Cycling performance under a low current density of 200 mA g^{-1} (c), long-term lithium-ion storage capability under a current density of 1000 mA g^{-1} (d), and rate capability (e) of the Fe_3O_4 QDs@C/RGO composite.

account for a continuous capacity increase in the following cycles.

The rate capability performance of the Fe_3O_4 QDs@C/RGO electrode is shown in Fig. 4e. The average reversible capacities at current densities of 100, 200, 500, 1000, 2000, 5000, and 10 000 mA g^{-1} are 1336.84, 1216.56, 1076.31, 980.17, 886.00, 723.83, and 607.12 mA h g^{-1}, respectively. The electrode provides a high reversible capacity of about 1179.44 mA h g^{-1} when the current density is restored to the initial current density of 100 mA g^{-1}, and the capacity is gradually increased in subsequent cycles. The unique structure with Fe_3O_4 QDs immobilized by pyrolytic carbon and RGO skeleton offers sufficient electric and ionic pathways at high current rate and maintains superior structural stability. Therefore, the extraordinary rate capability of the Fe_3O_4 QDs@C/RGO electrode can be achieved.

A brief comparison of the synthesis method, reversible capacity, cycle life, and current density of Fe_3O_4-based anode materials is listed in Table 1.[58–65] The Fe_3O_4 QDs@C/RGO composite electrode has the obvious merits of high reversible capacity, excellent long-time cycling stability as well as a facile synthesis method.

As presented in Fig. 5a, the dynamic behavior at different scan rates was tested to further investigate the lithium storage mechanism and diffusion kinetics of the Fe_3O_4 QDs@C/RGO electrode. The seven CV curves maintain similar properties and the peak current increases when experiencing a gradual increase in scan rate. These results indicate that the Fe_3O_4 QDs@C/RGO electrode has excellent electrochemical reversibility performance. The electrochemical reaction can be evaluated according to the relationship between peak current (i) and scanning rate (v), which follows the power-law relationship: $i = av^b$, where a and b are fitting parameters. The value of b varies between 0.5 (diffusion-controlled process) and 1.0 (capacitive-controlled process), which is obtained from the slope of a ln(i)–ln(v) plot.[66,67] As shown in Fig. 5b, the b values of the Fe_3O_4 QDs@C/RGO electrode can be calculated to be 0.714 and 0.855 for the cathode and anode peaks, which indicates the coexistence of diffusion-controlled processes and pseudocapacitive processes during the electrochemical reaction. The reversible capacity contribution can be calculated by the formula: $i = k_1v + k_2v^{1/2}$, where k_1v and $k_2v^{1/2}$ correspond to the pseudocapacitive contribution and diffusion-controlled contribution, respectively.[68,69] Fig. 5c shows the results of quantitatively analyzing the capacitive contribution of the Fe_3O_4 QDs@C/RGO electrode. At a scan rate of 1.0 mV s^{-1}, the pseudocapacitive contribution rate illustrated by the pink region is about 75% of the total lithium-ion storage capacity. In Fig. 5d, with the increase of scan rate from 0.1 mV s^{-1} to 1.0 mV s^{-1}, the pseudocapacitive contribution rate increases to 49.0%, 55.3%, 59.6%, 63.0%, 65.3%, 71.4%, and 75.0%, respectively. The pseudocapacitive behavior of the Fe_3O_4 QDs@C/RGO electrode is another factor for the inspiring rate performance.[70,71]

The galvanostatic intermittent titration technique (GITT) is regarded as an efficient method to explore the Li^+ diffusion coefficients (D_{Li^+}) for the Fe_3O_4 QDs@C/RGO composite electrode. Fig. 6a presents the GITT curves during the first cycle (tested at 200 mA g^{-1}, pulse time 20 min, relaxation time 30 min). According to Fick's second law of diffusion, D_{Li^+} is calculated as follows: $D_{Li^+} = \frac{4}{\pi\tau}\left(\frac{m_B V_m}{M_B A}\right)^2\left(\frac{\Delta E_S}{\Delta E_\tau}\right)^2$, in which τ is relaxation time (min), m_B, V_m, and M_B stand for the mass, molar volume and molecular weight of the electrode material, respectively, A represents the area of the electrode plate, ΔE_S is the steady-state potential change *via* the current pulse, and ΔE_τ is the potential change in the current pulse after subtracting the IR drop.[72–74] Fig. 6b illustrates a magnified single GITT curve, which is specifically marked with the parameters of τ, ΔE_S and ΔE_τ. As shown in Fig. 6c and d, the average value of D_{Li^+} reached 2.4×10^{-8} cm^2 s^{-1} and 2.6×10^{-8} cm^2 s^{-1} during the lithiation and delithiation processes, respectively. This result implies that an optimized Li^+ diffusion condition has been created in the Fe_3O_4 QDs@C/RGO composite.

FESEM images of the Fe_3O_4 QDs@C/RGO electrode before and after 100 cycles are employed to investigate the morphological and structural evolution. From Fig. 7a and b, it can be seen that the overall microstructure of the Fe_3O_4 QDs@C/RGO composite has been largely maintained after electrode preparation, indicating the good mechanical

Table 1 Comparison of the lithium-ion storage performance of carbon materials and carbonaceous matrix-supported Fe_3O_4 materials

Sample name	Synthesis method	Reversible capacity (mA h g^{-1})	Cycle life	Current density (mA g^{-1})	Year published
Fe_3O_4@C-500 (ref. 58)	Carbonization	718	500	200	2023
		190	2000	1000	
Fe_3O_4@HCS[59]	Hydrothermal carbonization	1050	250	100	2023
Fe_3O_4/C (ref. 60)	Carbonization	405.6	50	100	2023
Si-QDs/Fe_3O_4/rGO[61]	Self-assembly annealing	1367.1	80	100	2023
Fe_3O_4@void@N-doped C-5 (ref. 62)	Hydrothermal carbonization	1222	100	200	2022
Fe_3O_4@C (ref. 63)	Hydrothermal carbonization	291.7	300	1000	2022
PTA-700 (ref. 64)	Hydrothermal carbonization	535	100	100	2022
Fe_3O_4@CTP QDs[65]	Coprecipitation	810	200	100	2022
Fe_3O_4 QDs@C/RGO	Hydrothermal calcination	**738.03**	**300**	**1000**	**This work**
		1070.84	**150**	**200**	

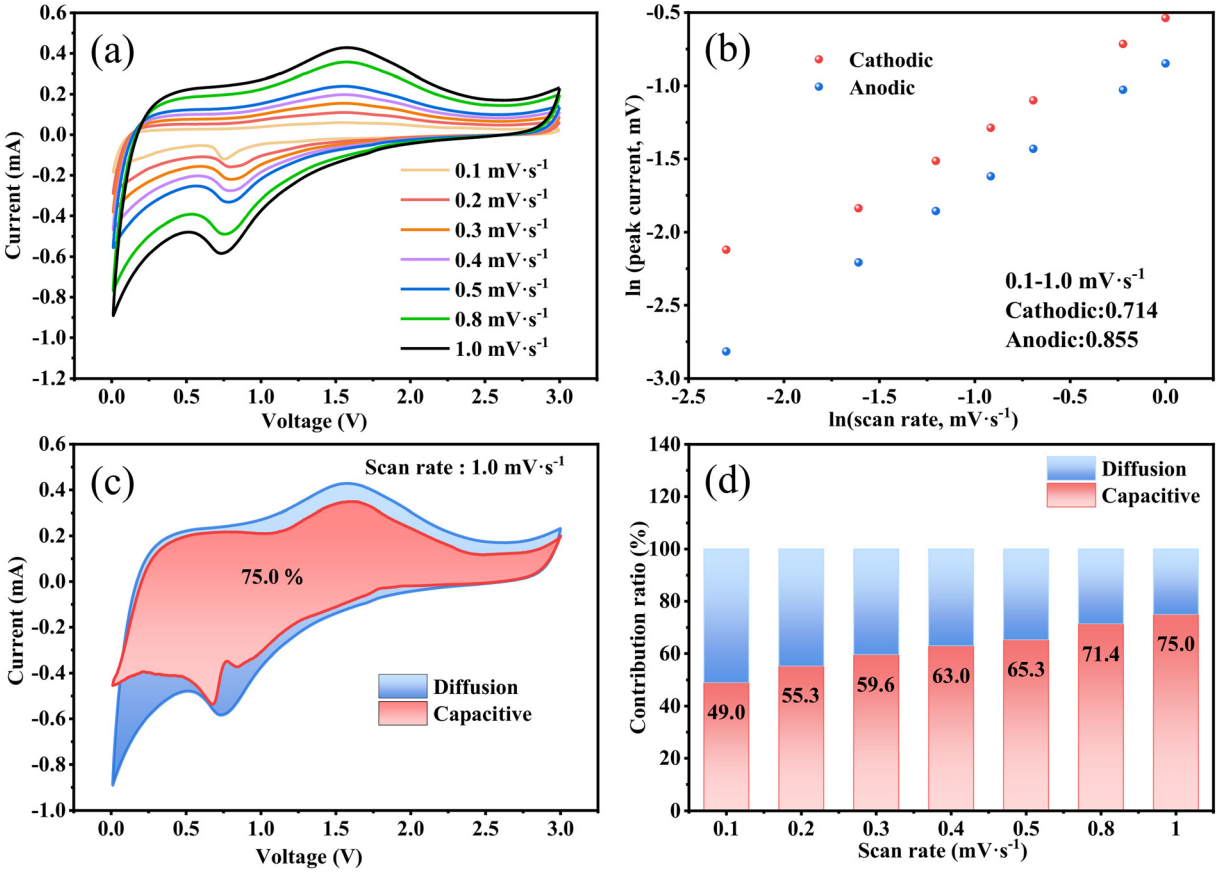


Fig. 5 CV curves of the Fe_3O_4 QDs@C/RGO composite electrode from 0.1 to 1.0 mV s^{-1} (a), the ln(*i*)–ln(*v*) plot (b), the capacitive contribution of the Fe_3O_4 QDs@C/RGO composite electrode at 1 mV s^{-1} (c) and the pseudocapacitive contribution ratios at different scanning rates (d).

strength of this composite sample. From Fig. 7c and d, an apparent smooth SEI film could be observed on the surface of the electrode after 100 cycles, while the electrode still maintains its integrity. These results well explain the good structural stability and excellent lithium storage properties of the Fe_3O_4 QDs@C/RGO electrode. EIS provides comprehensive interfacial electrochemical information, especially in the study of kinetic reaction mechanisms and the measurement of kinetic parameters of LIBs. The corresponding Nyquist plots of the Fe_3O_4 QDs@C/RGO electrode at different states are shown in Fig. 7e. In the equivalent circuit, constant phase angle element (CPE) is widely used in the equivalent circuit of alternating-current impedance to fit the experimental impedance data. The SEI film resistance (R_s) is the intersection of the curve starting point and the horizontal axis in the high-frequency region.[75] The charge transfer resistance (R_{ct}) is related to the diameter of the semicircle in the high-frequency and mid-frequency regions.[76] Warburg impedance (Z_w) corresponds to the slope of the line in the low-frequency region, reflecting the Li^+ diffusion in the electrode material.[77] According to the fitting results, it can be observed that the semicircle diameter obtained after 100 cycles for the Fe_3O_4 QDs@C/RGO composite electrode in the high- to medium-frequency region is significantly smaller than that before the cycling test, which illustrates the stable SEI formation at low potentials, better contact between the electrolyte and active material, and a significant reduction of charge transfer resistance after the activation of the Fe_3O_4 QDs@C/RGO composite.[78]

With a view to the practicality of the Fe_3O_4 QDs@C/RGO sample, full cells are configurated by integrating an Fe_3O_4 QDs@C/RGO anode and a commercial $LiFePO_4$ cathode. To guarantee the excellent electrochemical performance of full cells, the Fe_3O_4 QDs@C/RGO anode is electrochemically pre-lithiated to compensate for Li^+ consumption in the initial lithiation process before fabricating the full cells. Fig. 8a illustrates the charge/discharge profiles of the $LiFePO_4$ cathode material under a low current density of 200 mA g^{-1}, which exhibits a reversible plateau around 3.3 V. As shown in Fig. 8b, the reversible capacity reaches approximately 116.36 mA h g^{-1} after 100 cycles. Fig. 8c shows the galvanostatic charge/discharge voltage profiles of the Fe_3O_4 QDs@C/RGO//$LiFePO_4$ full cell between 0.01 and 3.7 V at a current density at 200 mA g^{-1}. The full cell exhibits initial lithiation and delithiation capacities of 1597.83 and 1268.54 mA h g^{-1}, and the corresponding coulombic efficiency is calculated to be 79.39%. Moreover, the shape of the curves for the subsequent cycles is largely maintained compared with the first cycle, indicating

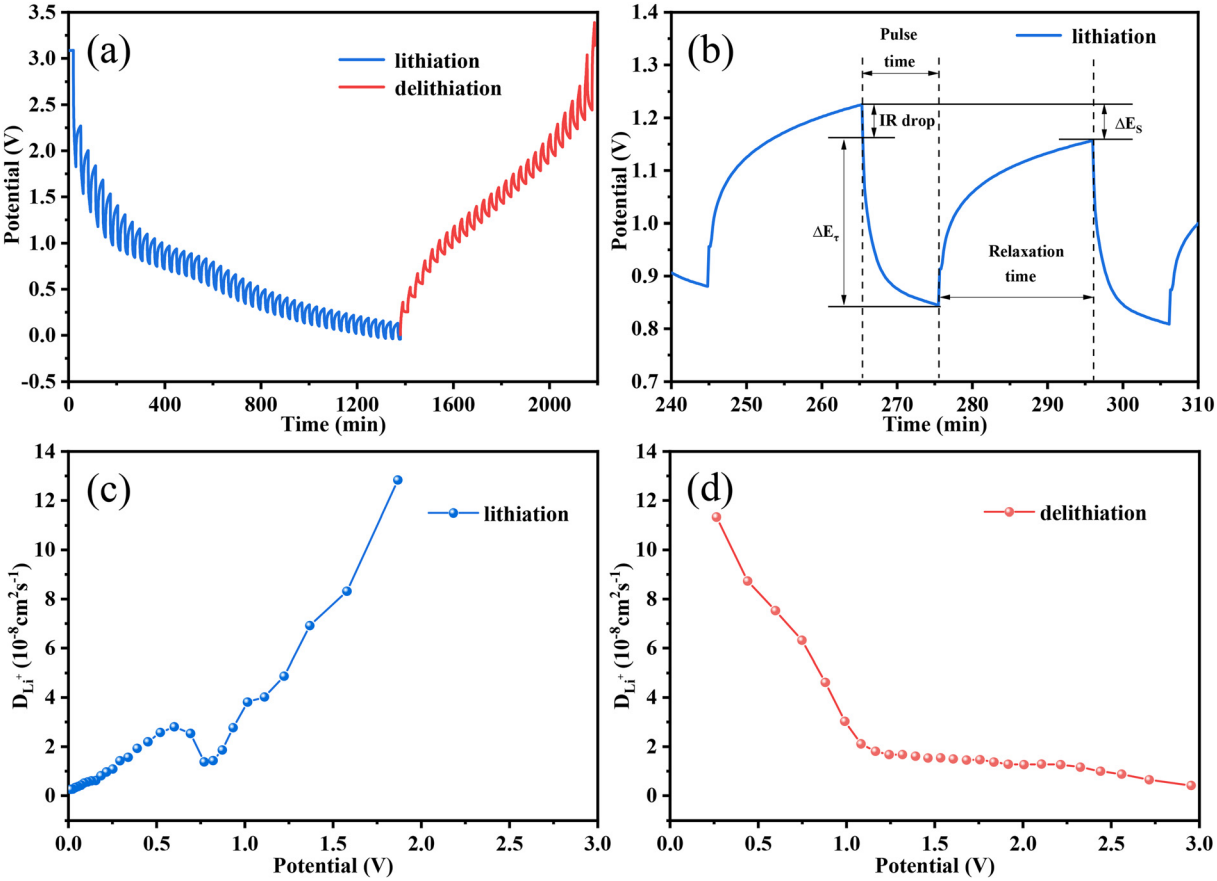
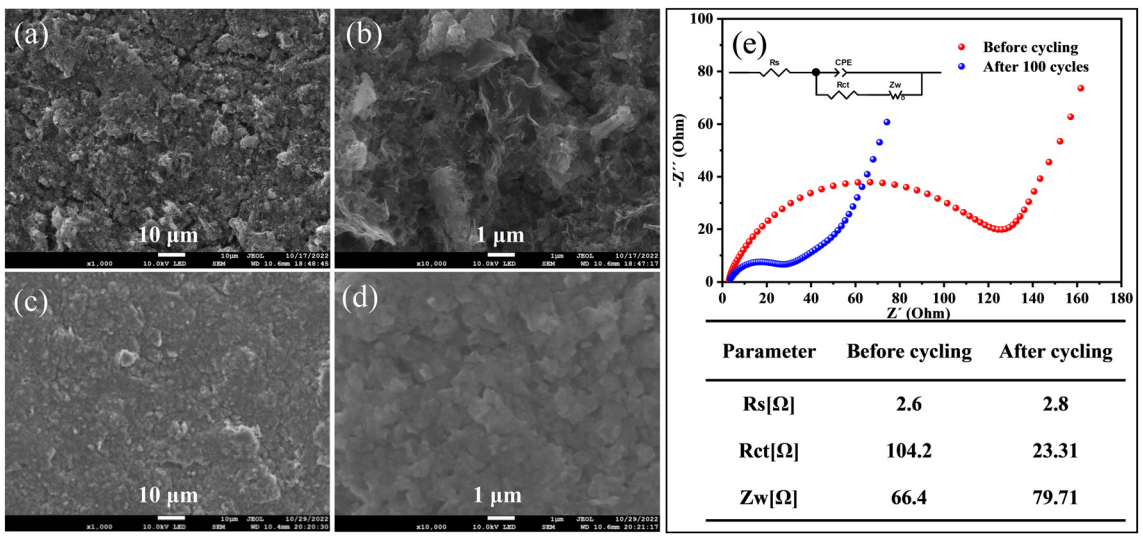


Fig. 6 GITT curves (a) and a profile for a single GITT test (b), and variation of the D_{Li^+} values during lithiation (c) and delithiation (d) processes of the Fe_3O_4 QDs@C/RGO electrode.



Parameter	Before cycling	After cycling
Rs[Ω]	2.6	2.8
Rct[Ω]	104.2	23.31
Zw[Ω]	66.4	79.71

Fig. 7 FESEM images of the Fe_3O_4 QDs@C/RGO electrode before cycling (a and b) and after 100 cycles (c and d) under various magnifications. Nyquist plots of the Fe_3O_4 QDs@C/RGO electrode before cycling and after 100 cycles with inset showing equivalent circuit and the corresponding fitted values of the parameters (e).

excellent structural stability for the composite over long-time cycling. From Fig. 8d, the full cell still exhibits a high reversible capacity of about 717.29 mA h g^{-1} after 150 cycles under a current density of 200 mA g^{-1}.

4. Conclusions

In a mild aqueous reaction system containing H_3BTC, TP, GO, and iron foils, an Fe-BTC/TP-RGO precursor sample with

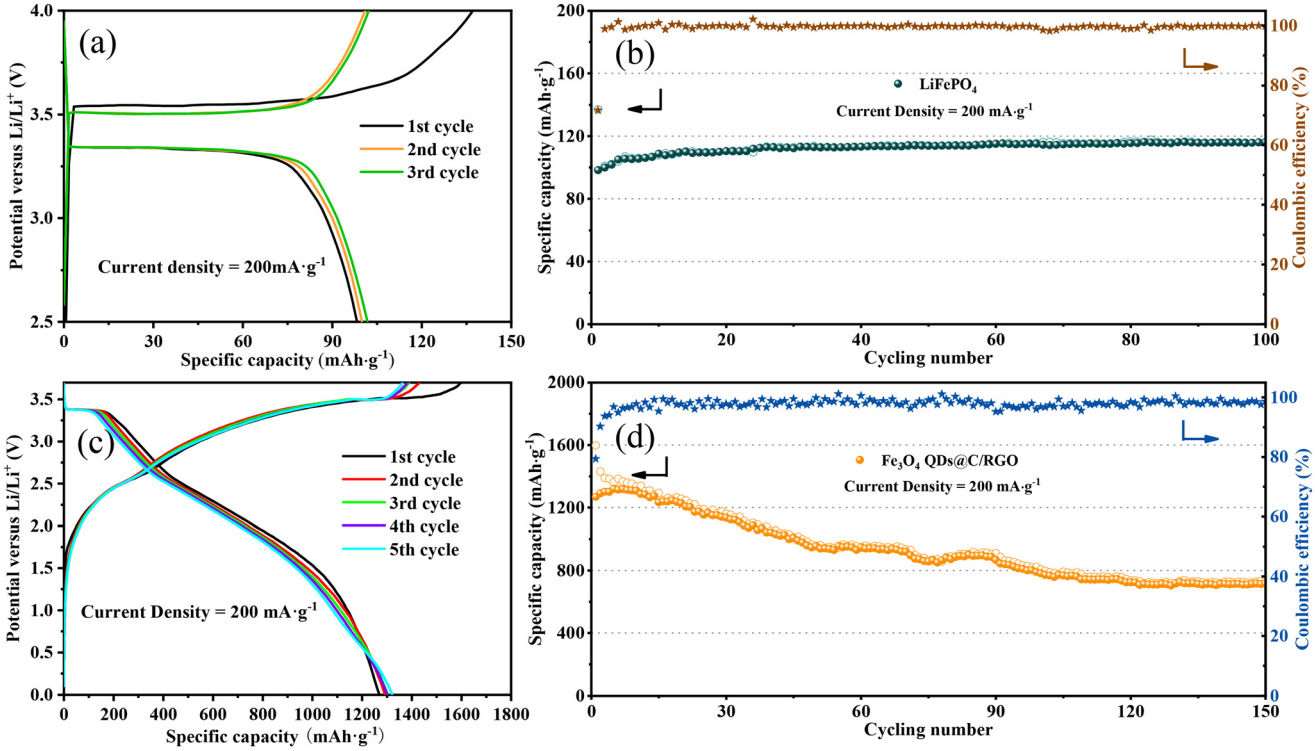


Fig. 8 Charge and discharge voltage profile (a) and cycling performance (b) of a commercial $LiFePO_4$ cathode at 200 mA g^{-1} in half cells. Charge and discharge voltage profile (c) and cycling performance (d) of the Fe_3O_4 QDs@C/RGO sample in Fe_3O_4 QDs@C/RGO//$LiFePO_4$ full cells at 200 mA g^{-1}.

improved microstructure is rationally engineered. Due to the modulation of the TP molecules, the spontaneous restacking of the RGO sheets has been effectively inhibited and the size growth of the Fe-BTC MOF domains has been significantly controlled. After a subsequent thermal treatment, the Fe_3O_4 QDs@C/RGO composite material is engineered, which has a unique structure with Fe_3O_4 QDs immobilized by pyrolytic carbon and RGO skeleton. The Fe_3O_4 QDs@C/RGO composite electrode exhibits a more attractive lithium-ion storage capacity with a reversible capacity of 1070.84 mA h g^{-1} for the electrode after 100 cycles at a low-density current of 200 mA g^{-1}, and a high reversible capacity of about 738.03 mA h g^{-1} after the 300th cycle at a high-density current of 1000 mA g^{-1}. Moreover, the Fe_3O_4 QDs@C/RGO sample could still have a high discharge capacity of 717.29 mA h g^{-1} after 150 cycles in Fe_3O_4 QDs@C/RGO//$LiFePO_4$ full cells between 0.01 and 3.7 V at 200 mA g^{-1}. The sample synthesis protocol is facile and eco-friendly, and the synthesized Fe_3O_4 QDs@C/RGO sample has good potential in wider fields.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was financially supported by the Thousand Talents Plan, the World-Class University and Discipline, the Taishan Scholar's Advantageous and Distinctive Discipline Program and the World-Class Discipline Program of Shandong Province.

References

1 F. Wu, J. Maier and Y. Yu, *Chem. Soc. Rev.*, 2020, **49**, 1569–1614.
2 J. Xu, X. Cai, S. Cai, Y. Shao, C. Hu, S. Lu and S. Ding, *Energy Environ. Sci.*, 2023, **0**, e12450.
3 J. Li, J. Fleetwood, W. B. Hawley and W. Kays, *Chem. Rev.*, 2022, **122**, 903–956.
4 L. Zhang, C. Zhu, S. Yu, D. Ge and H. Zhou, *J. Energy Chem.*, 2022, **66**, 260–294.
5 H. Cheng, J. G. Shapter, Y. Li and G. Gao, *J. Energy Chem.*, 2021, **57**, 451–468.
6 C. Wang, C. Yang and Z. Zheng, *Adv. Sci.*, 2022, **9**, e2105213.
7 C. Hu, Y. J. Hu, A. P. Chen, X. Z. Duan, H. Jiang and C. Z. Li, *Engineering*, 2022, **18**, 154–160.
8 Y. Liu, C. Hu, L. Chen, Y. J. Hu, H. Jiang and C. Z. Li, *J. Energy Chem.*, 2022, **69**, 450–455.
9 C. Hu, L. Chen, Y. J. Hu, A. P. Chen, L. Chen, H. Jiang and C. Z. Li, *Adv. Mater.*, 2021, **33**, 2103558.
10 J. P. Xie, J. L. Li, X. D. Li, H. Lei, W. C. Zhuo, X. B. Li, G. Hong, K. N. Hui, L. K. Pan and W. J. Mai, *CCS Chem.*, 2021, **3**, 791–799.
11 Z. Yi, D. L. Fang, W. Q. Zhang, J. Tian, S. M. Chen, J. B. Liang, N. Lin and Y. T. Qian, *CCS Chem.*, 2021, **3**, 1306–1315.

12 N. Wang, Y. Y. Liu, Z. X. Shi, Z. L. Yu, H. Y. Duan, S. Fang, J. Y. Yang and X. M. Wang, *Rare Met.*, 2021, **41**, 438–447.
13 S. Fang, D. Bresser and S. Passerini, *Adv. Energy Mater.*, 2019, **10**, 1902485.
14 Y. Yang, W. Yuan, X. Zhang, C. Wang, Y. Yuan, Y. Huang, Y. Ye, Z. Qiu and Y. Tang, *Renewable Sustainable Energy Rev.*, 2020, **127**, 109884.
15 R. Puttaswamy, R. K. Pai and D. Ghosh, *J. Mater. Chem. A*, 2022, **10**, 508–553.
16 Y. Yu, T. Ma and H. Huang, *Adv. Funct. Mater.*, 2023, **33**, 2213770.
17 J. F. S. Fernando, C. Zhang, K. L. Firestein, J. Y. Nerkar and D. V. Golberg, *J. Mater. Chem. A*, 2019, **7**, 8460–8471.
18 C. X. Peng, B. D. Chen, Y. Qin, S. H. Yang, C. Z. Li, Y. H. Zuo, S. Y. Liu and J. H. Yang, *ACS Nano*, 2012, **6**, 1074–1081.
19 J. Y. Wu, H. Lin, D. J. Moss, K. P. Loh and B. H. Jia, *Nat. Rev. Chem.*, 2023, **7**, 162–183.
20 F. Wang, W. Fang, X. Ming, Y. Liu, Z. Xu and C. Gao, *Appl. Phys. Rev.*, 2023, **10**, 011311.
21 R. C. K. Reddy, J. Lin, Y. Chen, C. Zeng, X. Lin, Y. Cai and C.-Y. Su, *Coord. Chem. Rev.*, 2020, **420**, 213434.
22 R. A. Bajwa, U. Farooq, S. Ullah, M. Salman, S. Haider and R. Hussain, *J. Energy Storage*, 2023, **72**, 108708.
23 M. H. Shen and H. L. Ma, *Coord. Chem. Rev.*, 2022, **470**, 214715.
24 Y. Zhang, Y. F. Liu, D. Wang, J. C. Liu, J. W. Zhao and L. J. Chen, *Polyoxometalates*, 2023, **2**, 9140017.
25 Y. An, L. L. Wang, W. Y. Jiang, X. L. Lv, G. Q. Yuan, X. X. Hang and H. Pang, *Polyoxometalates*, 2023, **2**, 9140030.
26 R. Jia, R. Zhang, L. Yu, X. Kong, S. Bao, M. Tu, X. Liu and B. Xu, *J. Colloid Interface Sci.*, 2023, **630**, 86–98.
27 Y. Wang, Z. Shi and J. Yin, *ACS Appl. Mater. Interfaces*, 2011, **3**, 1127–1133.
28 O. Akhavan, M. Kalaee, Z. S. Alavi, S. M. A. Ghiasi and A. Esfandiar, *Carbon*, 2012, **50**, 3015–3025.
29 X. Weng, J. Wu, L. Ma, G. Owens and Z. Chen, *Chem. Eng. J.*, 2019, **359**, 976–981.
30 B. Xu, J. Zhang, Y. Gu, Z. Zhang, W. Al Abdulla, N. A. Kumar and X. S. Zhao, *Electrochim. Acta*, 2016, **212**, 473–480.
31 Y. Yang, Y. Bai, F. Zhao, E. Yao, J. Yi, C. Xuan and S. Chen, *RSC Adv.*, 2016, **6**, 67308–67314.
32 T. A. Vu, G. H. Le, H. T. Vu, K. T. Nguyen, T. T. T. Quan, Q. K. Nguyen, H. T. K. Tran, P. T. Dang, L. D. Vu and G. D. Lee, *Mater. Res. Express*, 2017, **4**, 035038.
33 H. T. Vu, M. B. Nguyen, T. M. Vu, G. H. Le, T. T. T. Pham, T. D. Nguyen and T. A. Vu, *Top. Catal.*, 2020, **63**, 1046–1055.
34 B. Xu, L. Yu, X. Zhao, H. Wang, C. Wang, L. Y. Zhang and G. Wu, *J. Colloid Interface Sci.*, 2021, **584**, 827–837.
35 C. Xiao, H. Li, Y. Zhao, X. Zhang and X. Wang, *J. Environ. Manage.*, 2020, **275**, 111262.
36 X. Kong, L. Shan, R. Zhang, S. Bao, M. Tu, R. Jia, L. Yu, H. Li and B. Xu, *J. Colloid Interface Sci.*, 2022, **628**, 1–13.
37 P. Pachfule, D. Shinde, M. Majumder and Q. Xu, *Nat. Chem.*, 2016, **8**, 718–724.
38 D. Graf, F. Molitor, K. Ensslin, C. Stampfer, A. Jungen, C. Hierold and L. Wirtz, *Nano Lett.*, 2007, **7**, 238–242.
39 Z. Shen, H. Xing, Y. Zhu, X. Ji, Z. Liu and L. Wang, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 13896–13904.
40 B. Xu, X. Guan, L. Y. Zhang, X. Liu, Z. Jiao, X. Liu, X. Hu and X. S. Zhao, *J. Mater. Chem. A*, 2018, **6**, 4048–4054.
41 Y. Cui, W. Feng, W. Liu, J. Li, Y. Zhang, Y. Du, M. Li, W. Huang, H. Wang and S. Liu, *Nanoscale*, 2020, **12**, 10816–10826.
42 Y. Liu, D. He, Q. Tan, Q. Wan, K. Han, Z. Liu, P. Li, F. An and X. Qu, *J. Mater. Chem. A*, 2019, **7**, 19430–19441.
43 Q. Wu, R. Yu, Z. Zhou, H. Liu and R. Jiang, *Langmuir*, 2021, **37**, 785–792.
44 Y. Yuan, Z. Kong, L. Qiao, Z. Xu, Z. Wang, X. Teng, Y. Dong, X. Liu, A. Fu, Y. Li and H. Li, *ChemElectroChem*, 2021, **8**, 4480–4489.
45 Y. Yin, C. Ma, W. D. Cai, W. M. Qiao, L. C. Ling and J. T. Wang, *ChemNanoMat*, 2022, **8**, e202100490.
46 X. Meng, Y. Xu, X. Sun, J. Wang, L. Xiong, X. Du and S. Mao, *J. Mater. Chem. A*, 2015, **3**, 12938–12946.
47 Y. He, L. Huang, J.-S. Cai, X.-M. Zheng and S.-G. Sun, *Electrochim. Acta*, 2010, **55**, 1140–1144.
48 Y. Wang, L. Chen, H. Liu, Z. Xiong, L. Zhao, S. Liu, C. Huang and Y. Zhao, *Chem. Eng. J.*, 2019, **356**, 746–755.
49 Y. Cui, W. Liu, Y. Lyu, Y. Zhang, H. Wang, Y. Liu and D. Li, *J. Mater. Chem. A*, 2018, **6**, 18276–18285.
50 J. Song, Y. Ji, Y. Li, X. Lu, W. Ren, Q. Tian, J. Chen and L. Yang, *Ceram. Int.*, 2021, **47**, 26092–26099.
51 Y. Ma, J. Huang, L. Lin, Q. Xie, M. Yan, B. Qu, L. Wang, L. Mai and D. Peng, *J. Power Sources*, 2017, **365**, 98–108.
52 J. Cheng, Y. Zhou, Y. Wu, Y. Pan, X. Luo, Y. Huang, Y. Zhou, L. Zhu and Z. Yuan, *J. Alloys Compd.*, 2023, **930**, 167429.
53 D. Yoon, J. Hwang, W. Chang and J. Kim, *Chem. Eng. J.*, 2017, **317**, 890–900.
54 R. Zhang, S. Bao, Q. Tan, B. Li, C. Wang, L. Shan, C. Wang and B. Xu, *J. Colloid Interface Sci.*, 2021, **600**, 602–612.
55 Y. Yan, X. Lu, Y. Li, J. Song, Q. Tian, L. Yang and Z. Sui, *J. Alloys Compd.*, 2022, **899**, 163342.
56 R. Zhang, C. Lv, S. Bao, J. Gao, Y. Xie, F. Zheng, X. Liu, Y. Wen and B. Xu, *J. Colloid Interface Sci.*, 2022, **628**, 154–165.
57 X. Meng, J. Huang, Y. Bian, H. Du, Y. Xu, S. Zhu, Q. Li, M. Chen and M.-C. Lin, *J. Solid State Chem.*, 2021, **303**, 122456.
58 M. Li, W. Ma, F. Tan, B. Yu, G. Cheng, H. Gao and Z. Zhang, *J. Power Sources*, 2023, **574**, 233146.
59 L. Singer, W. Kukułka, E. Thauer, N. Gräßler, A. Asyuda, M. Zharnikov, E. Mijowska and R. Klingeler, *Electrochim. Acta*, 2023, **448**, 142155.
60 W. Yodying, T. Sarakonsri, N. Ratsameetammajak, K. Khunpakdee, M. Haruta and T. Autthawong, *Crystals*, 2023, **13**, 280.
61 W. Hu, K. He, S. Wu, T. Chen, X. Yu, Y. Liang, M. Zheng, Y. Xiao, H. Dong, Y. Liu and H. Hu, *J. Alloys Compd.*, 2023, **943**, 168947.
62 J. Wang, Q. Hu, W. Hu, W. Zhu, Y. Wei, K. Pan, M. Zheng and H. Pang, *Molecules*, 2022, **27**, 396.
63 J. Huang, Q. Dai, C. Cui, H. Ren, X. Lu, Y. Hong and S. Woo Joo, *J. Electroanal. Chem.*, 2022, **918**, 116508.

64 Q. Wei, H. Zhu, S. Yu, G. Xu, J. Yin, J. Tong, T. Chen, X. He, P. Guo, H. Jiang, J. Li and Y. Wang, *Appl. Surf. Sci.*, 2023, **608**, 155093.
65 F. Shi, Q. Liu, Z. Jin, G. Huang, B. Xing, J. Jia and C. Zhang, *J. Alloys Compd.*, 2022, **890**, 161911.
66 V. Augustyn, J. Come, M. A. Lowe, J. W. Kim, P. L. Taberna, S. H. Tolbert, H. D. Abruna, P. Simon and B. Dunn, *Nat. Mater.*, 2013, **12**, 518–522.
67 Q. Chen, W. Zhong, J. Zhang, C. Gao, W. Liu, G. Li and M. Ren, *J. Alloys Compd.*, 2019, **772**, 557–564.
68 V. Augustyn, P. Simon and B. Dunn, *Energy Environ. Sci.*, 2014, **7**, 1597.
69 Z. Liu, L. Liu, Z. Zhao, J. He, S. Wang and C. Xiong, *Appl. Surf. Sci.*, 2020, **526**, 146639.
70 J. Song, Y. Li, X. Lu, W. Zhang, Y. Xiang, J. Chen and Q. Tian, *Appl. Surf. Sci.*, 2022, **604**, 154502.
71 B. Yin, X. Cao, A. Pan, Z. Luo, S. Dinesh, J. Lin, Y. Tang, S. Liang and G. Cao, *Adv. Sci.*, 2018, **5**, 1800829.
72 C. Lv, C. Lin and X. S. Zhao, *Adv. Energy Mater.*, 2021, **12**, 2102550.
73 H. Zhang, P. Zong, M. Chen, H. Jin, Y. Bai, S. Li, F. Ma, H. Xu and K. Lian, *ACS Nano*, 2019, **13**, 3054–3062.
74 Y. Cheng, S. Wang, L. Zhou, L. Chang, W. Liu, D. Yin, Z. Yi and L. Wang, *Small*, 2020, **16**, e2000681.
75 R. Ding, J. Zhang, J. Qi, Z. Li, C. Wang and M. Chen, *ACS Appl. Mater. Interfaces*, 2018, **10**, 13470–13478.
76 Y. Xie, Y. Qiu, L. Tian, T. Liu and X. Su, *J. Alloys Compd.*, 2022, **894**, 162384.
77 H. Li, J. Wang, Y. Li, Y. Zhao, Y. Tian, I. Kurmanbayeva and Z. Bakenov, *J. Electroanal. Chem.*, 2019, **847**, 113240.
78 Y. Zhao, X. Zhai, D. Yan, C. Ding, N. Wu, D. Su, Y. Zhao, H. Zhou, X. Zhao, J. Li and H. Jin, *Electrochim. Acta*, 2017, **243**, 18–25.