



Cite this: *Phys. Chem. Chem. Phys.*,
2021, **23**, 26336

Time- and momentum-resolved image-potential states of 2H-MoS₂ surface†

Jianyi Liu, Xiang Jiang, Xintong Li, Xiaochuan Ma, Xia Sun,  Qijing Zheng,
Xuefeng Cui,* Shijing Tan,  Jin Zhao * and Bing Wang *

Rydberg-like image potential states (IPs) form special series surface states on metal and semiconducting surfaces. Here, using time-resolved and momentum-resolved multi-photon photoemission (*mPPE*), we measured the energy positions, band dispersion, and carrier lifetimes of IPs at the 2H-MoS₂ surface. The energy minima of the IPs ($n = 1$ and 2) were located at 0.77 and 0.21 eV below the vacuum level. In addition, the effective masses of these two IPs are close to the rest mass of the free electron, clearly showing nearly-free-electron character. These properties suggest a good screening effect in the MoS₂ parallel to the surface. The multi-photon resonances between the valence band and IP ($n = 1$) are observed, showing a k_{\parallel} -momentum-dependent behavior. Our time-resolved *mPPE* measurements show that the lifetime of photoexcited electrons in the IP ($n = 1$) is about 33 fs.

Received 31st July 2021,
Accepted 28th October 2021

DOI: 10.1039/d1cp03527d

rsc.li/pccp

1. Introduction

Rydberg-like image potential states (IPs) form a special class of electronic states at different surfaces where they are bound by their image charges through the long-range Coulomb interaction. IPs at metal surfaces have been intensively studied since 1985,¹ and they have also been observed at semiconducting surfaces.^{2,3} As a special series of surface states, IPs are extremely sensitive to surface properties. They are bound by the electrostatic Coulomb potential and reside few angstroms outside the surface, making them particularly alert to changes in the surface electrostatic potential. For example, with molecules or atoms adsorbed on the surface, interfacial charge transfer or reorienting molecular dipoles in the adsorbate layer can change the surface dipoles, which will affect the energy positions of the IPs.^{4–6} Furthermore, the molecular layer on metal surfaces can be considered as a quantum well, which confines the image wavefunction by the thickness of the adsorbate layer.^{7,8} Moreover, the IPs may couple strongly to the other electronic bands through photoexcitation. As a series of unoccupied states, they have been reported to be mediated states for charge transfer dynamics at different interfaces.^{9–12} Therefore, the measurement of IPs, especially the dynamics of IPs, is of great importance to understand the dynamics on the surface.

Compared to traditional metal surfaces, the newly developed two-dimensional (2D) material family brings new chances for novel optoelectronic devices, in which the charge transfer dynamics play a key role.^{13–15} The measurement of IP dynamics on 2D materials will provide an approach to investigate the charge transfer dynamics at the surfaces and interfaces of 2D materials. In addition, unlike those electronic states that usually have a density distribution localized around the nuclei, the maxima of IPs are located away from the atomic centers; namely, they have nonnuclear density maxima, showing diffusive character and forming energy bands with nearly-free-electron (NFE) character.^{16–18} Such NFE-like IPs are expected to exist in low-dimensional nanomaterials. For example, the existence of IPs in two-dimensional (2D) graphene was theoretically predicted.^{19–21} When 2D graphene is wrapped to form a one-dimensional (1D) carbon nanotube or zero-dimensional (0D) fullerenes, the IPs can transfer into the 1D NFE states or 0D superatom molecular orbitals.^{22–24} All these states have nonnuclear density maxima, which were proposed to have small electron–phonon interactions and to provide unique conduction channels.²⁵ Therefore, it is also of great interest to understand the properties of IPs in low-dimensional materials,^{21,26} especially in the newly developed 2D-materials family.²⁷

Transition metal dichalcogenides (TMDs) form one of the most intensively studied 2D semiconductor families, which have important applications in spintronics and optoelectronics.^{14,15,28–30} Extensive research efforts have been performed on their electronic structures, charge carrier dynamics and light–matter interaction.^{31–35} The energy positions and band dispersion of IP are related to 2D screening properties. The electron can be excited

Hefei National Laboratory for Physical Sciences at the Microscale and Synergetic Innovation Center of Quantum Information & Quantum Physics, CAS Key Laboratory of Strongly-Coupled Quantum Matter Physics, and Department of Physics, University of Science and Technology of China, Hefei, Anhui 230026, China. E-mail: xf cui@ustc.edu.cn, zhaojin@ustc.edu.cn, bwang@ustc.edu.cn
† Electronic supplementary information (ESI) available. See DOI: 10.1039/d1cp03527d

into the IPS by photons. Such photoexcited electrons with a certain lifetime on IPS can scatter with phonons and other charge carriers during its relaxation, which will affect the photoexcited charge carrier dynamics on TMD surface.³³

MoS₂ is a representative material in various TMDs. Bulk 2H-MoS₂ is an indirect gap semiconductor with a bandgap of about 1.29 eV,³² which is different from monolayer MoS₂ with a direct bandgap of about 1.8 eV.³⁶ Despite their difference in the bandgaps, understanding the electron dynamics of the IPSs in bulk 2H-MoS₂ should be an important step for further understanding of other competing dynamical properties in less-layered MoS₂ and even TMD materials.³⁷ Many studies have been carried out on the dynamics of excited states in conduction bands or intervalley scattering in MoS₂.^{31,33–35,38} However, to our best knowledge, we are unaware of any report on the dynamical properties of trapped electrons in MoS₂ IPSs, and it is even unclear if there exist resolvable IPSs at MoS₂ surfaces.

In this work, we, for the first time investigate the hot electron dynamics of IPSs at bulk 2H-MoS₂ surface by means of time-resolved and momentum-resolved multi-photon photoemission (TR-MR-*m*PPE) spectroscopy, with $m = 3, 4,$ and 5 using varying the photon energies. Scanning tunneling microscopy (STM) and conventional angle-resolved photoemission spectroscopy (ARPES) were also used for characterizing the surface morphology and the valence band structure of the 2H-MoS₂ surface. In our results, the observed parabolic dispersion of the states near the vacuum level of the 2H-MoS₂ surface can be well described by a series of Rydberg-like image potential states, in good agreement with the features of IPSs. The observed effective masses of electrons in the IPSs are very close to the rest mass of a free electron. These features are well supported by our theoretical calculations. Multiphoton excitation processes are identified, and the momentum-dependent resonances between IPS and the valence band maximum (VBM) are observed. The lifetime of the 2H-MoS₂ IPS was also measured.

2. Experimental and theoretical

The experiments were carried out in an ultrahigh vacuum (UHV) system equipped with a low-temperature scanning tunneling microscope (LT-STM, Omicron), and a TR-MR-*m*PPE system, similar to the system developed by Reutzler *et al.*,³⁹ that recorded spectra using a hemispherical electron energy analyzer (SPECS PHOIBOS 150) with a 2D delay-line detector (DLD). The base pressure of the system was better than 2×10^{-10} mbar. The photo-excitation source for the *m*PPE measurements was a noncollinear parametric amplifier (NOPA) system pumped by a Clark MXR Impulse fiber laser. The NOPA system was performed at a power of about 10 μ J pulse⁻¹ with a repetition rate of 500 kHz and a tunable wavelength in the range of 340–900 nm (pulse width \approx 30 fs after compensation with a compressor). The *p*-polarized laser light with an incident angle of 45° from the surface normal was used, with a focused

beam size of about 50 μ m at the sample surface. In the interferometric measurements, identical pump–probe pulse pairs were generated using a Mach-Zehnder interferometer,⁴⁰ and the delay time was scanned using a piezo driver that electrically varies the path length of one arm. One output of the interferometer was directly focused onto the sample as the *m*PPE excitation source, while the other one was transmitted through a monochromator to generate interference fringes that calibrate the pump–probe delay time.¹² More than 200 pump–probe scans were accumulated to improve the counting statistics.

Commercially purchased 2H-MoS₂ single crystal (HQ Graphene, Netherlands) was freshly cleaved in the UHV chamber, and then was further cleaned by annealing at about 450 K for 20 minutes. The cleanliness and the surface quality were characterized *in situ* using STM before *m*PPE measurements. The band structure of the 2H-MoS₂ surface was also measured *ex situ* using an ARPES (VG SCIENTIA, DA30L). Before the ARPES measurements, the sample was treated according to the similar procedure used in the sample for the *m*PPE measurements. In the ARPES measurements, a high brightness helium discharge lamp of I_z emission line (21.22 eV) was used. During ARPES measurements, the samples were kept at 90 K.

Density functional theory (DFT) calculations were performed with the Vienna *Ab initio* Simulation Package⁴¹ (VASP) using Perdew–Burke–Ernzerhof⁴² (PBE) exchange–correlation functional, the projector-augmented wave (PAW) method and the D3 van der Waals correction by Grimme *et al.*⁴³ The kinetic energy cutoff was set at 400 eV for the plane-wave-basis setting and the Brillouin zone sampled with a $24 \times 24 \times 1$ Γ -centered k -point mesh. There is a more than 50 Å vacuum portion above the MoS₂ surface with a 10-layer MoS₂ bulk region.

3. Results and discussion

Fig. 1(a) schematically shows the measurements using a visible pump and probe laser to record TR-MR-*m*PPE spectra of bulk 2H-MoS₂ surface, that is, using 3D spectroscopy.³⁹ The high-quality surface of the 2H-MoS₂ sample was characterized by the atomically resolved STM images, as shown in Fig. 1(b). The observed lattice parameter of 0.34 ± 0.01 nm and the angle of 60° between the lattice vectors is consistent with the centrosymmetric phase structure of the 2H-MoS₂ surface along the (0001) direction.^{44,45}

Fig. 1(c) shows a typical energy-momentum (E - k_{\parallel}) map of the band structure measured at an excitation photon energy of 1.61 eV at zero pump–probe delay time. The final-state (E_{final}) energies recorded by the analyzer are referenced with respect to the Fermi energy ($E_{\text{f}} = 0$). Fig. 1(d) is the integrated line trace at $k_{\parallel} = 0$. The low-energy cutoff corresponds to the vacuum level (E_{vac}) through the *m*PPE process, which determines the work function (Φ) of the bulk 2H-MoS₂ surface by $\Phi = E_{\text{vac}} - E_{\text{f}} = 4.98$ eV.⁴⁶ Three upward-dispersed bands are observed, which can be well fitted with the parabolic line shape (overlaid dashed red lines), with an approximate effective mass of

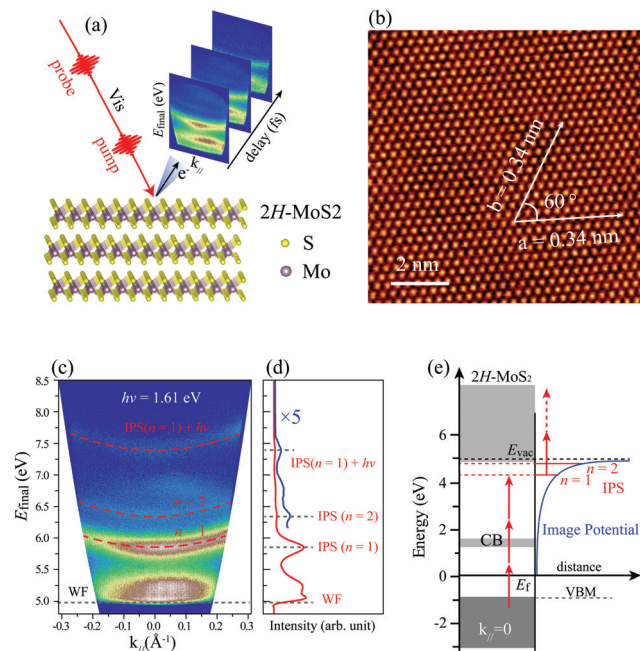


Fig. 1 (a) Schematic drawing to show the measurements of TR-MR-*mPPE* spectra for the bulk 2H-MoS₂ surface using the pump-probe method. (b) An STM image of the MoS₂ surface acquired at a sample bias of 1.0 V and the tunneling current of 140 pA. (c) Typical *mPPE* spectrum of 2H-MoS₂ surface measured using photon energy of 1.61 eV at zero pump-probe delay time. The dashed red lines are the fitting curves of the parabolic dispersed bands. (d) Integrated signals of near $k_{\parallel} = 0$ within $\Delta k_{\parallel} = \pm 0.03 \text{ \AA}^{-1}$. The blue line is a magnified segment at high-energy final states. The peak positions are labeled by WF (work function of MoS₂), Rydberg-like series $n = 1, 2$ (IPS), and IPS ($n = 1$) + $h\nu$. (e) Energy diagram to show the *mPPE* processes through the intermediate IPSs.

$m^* \approx m_e$, where m_e is the rest mass of the electron. On the basis of the parabolic dispersion of these bands in accordance with the band dispersion of IPSs, we tentatively attribute them to IPS ($n = 1$), IPS ($n = 2$), and IPS-related state (IPS ($n = 1$) + $h\nu$). Fig. 1(e) shows the energy diagram extracted from the experimental observations, where the horizontal red lines below the E_{vac} correspond to the IPS series $n = 1$ and 2, respectively.

To confirm our assignments, we performed a series of *mPPE* measurements by varying the excitation photon energy from 1.57 eV to 1.72 eV recorded at zero delay time (Fig. S1, ESI[†]). The corresponding line traces integrated near $k_{\parallel} = 0$ from each *mPPE* spectrum in Fig. S1 (ESI[†]) are plotted as a function of photon energy in Fig. 2(a). The extracted peak positions of IPS ($n = 1$) and IPS ($n = 2$) in the final-state energies are then plotted as a function of excitation photon energy in Fig. 2(b), which present explicit linear dependence. The slopes by linear fitting of the data in IPS ($n = 1$) and IPS ($n = 2$) are both around 1, which indicates that in both cases the emission takes place by absorbing one photon, while the multi-photon order for both states is 4.⁴⁷ Therefore, the energy positions of IPS ($n = 1$) and IPS ($n = 2$) at $k_{\parallel} = 0$ can be obtained by $E_n = E_{\text{final}} - h\nu$. The observed energy difference between IPS ($n = 1$) and IPS ($n = 2$) is ~ 0.56 eV. The high-energy state of IPS ($n = 1$) + $h\nu$ can be assigned to the replica of IPS ($n = 1$) in above-threshold

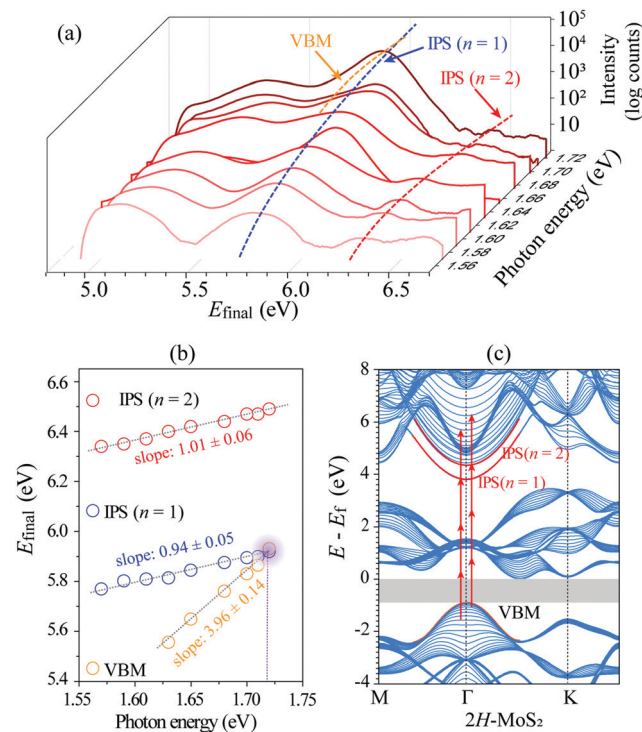


Fig. 2 (a) Plot of integrated signals at $k_{\parallel} = 0$ against photon energy and the final-state energy using a *p*-polarized light. (b) Peak positions in the final-state energies against photon energies. The dotted lines represent linear fitting of the data of IPS ($n = 1$) and IPS ($n = 2$), respectively. (c) Calculated band structure along $M-\Gamma-K$ for the 2H-MoS₂ surface. The two upward dispersed red parabolic curves denote the IPS ($n = 1$) and IPS ($n = 2$) band, and the lower downward parabolic curve denotes the valence band maximum (VBM). The arrowed red segments indicate photoemission transitions via the IPSs.

photoemission (ATP).^{21,48} The extracted values of $(E_n - E_f)$, $(E_{\text{vac}} - E_n)$, and m^* are summarized in Table 1. We also measured the distance-voltage (z - V) curves and dz/dV curves at different constant-current setpoints. As shown in Fig. S3 (ESI[†]), a series of setpoint-current-dependent steps or peaks are observed in the z - V or dz/dV curves. The peak labeled by $n = 1$ positioned below the E_{vac} by about 0.5–0.6 eV should correspond to the IPS ($n = 1$) in the *mPPE* measurements, where their energy difference can be ascribed to the Stark shift in the STM measurements.⁴⁹ Hence, the existence of the IPSs can also be supported.

Table 1 Measured and calculated energy positions of Rydberg-like states (relative to the E_f and E_{vac} , respectively) and effective masses m^* of the n th image potential state at the 2H-MoS₂ surface

		$E_n - E_f$ (eV)	$E_{\text{vac}} - E_n$ (eV)	m^*/m_e
TR- <i>mPPE</i>	IPS ($n = 1$)	4.21 ± 0.04	0.77 ± 0.02	1.05 ± 0.06
	IPS ($n = 2$)	4.77 ± 0.02	0.21 ± 0.02	0.99 ± 0.06
Hydrogenic model	IPS ($n = 1$)	0.66		
	IPS ($n = 2$)	0.16		
DFT	IPS ($n = 1$)	0.57	0.9	
	IPS ($n = 2$)	0.02	1.0	

On semiconducting surfaces, the IPS binding energy, namely, energy relative to E_{vac} can be estimated using a classical Rydberg-like series:^{3,17,50}

$$E_n = E_{\text{vac}} - \frac{0.85}{n^2} \left(\frac{\epsilon_r - 1}{\epsilon_r + 1} \right)^2 \text{ eV} \quad (1)$$

where n is the quantum number of the IPS and ϵ_r is the static dielectric constant.^{3,51} From our DFT calculations, the ϵ_r of MoS₂ is 15.6 in the direction parallel to the surface, which is consistent with the previous results.^{52,53} Based on this value, ($E_{\text{vac}} - E_n$) of IPSs ($n = 1$) and ($n = 2$) were calculated as 0.66 and 0.16 eV, respectively. These values are in agreement with the MR-*m*PPE measurements in Fig. 1(c), suggesting that the IPSs on MoS₂ can be described sufficiently by the classical hydrogenic model. The binding energies of IPSs on MoS₂ are comparable to the metal surfaces,⁵⁴ but slightly larger than that on wide-bandgap semiconductor NiO.³ It suggests that the MoS₂ has good screening properties analogous to metals in the direction parallel to the surface.

We also performed theoretical calculations on the band structure of the bulk 2H-MoS₂ based on DFT. Fig. 2(c) shows the projected band structure of 2H-MoS₂ at the $M-\Gamma-K$ plane calculated using a slab model consisting of a 5-layer MoS₂ using a tight-binding (TB) approach. It was observed that the calculated band structure in the VB is in good agreement with the experimentally measured band structure using the ARPES (Fig. S2, ESI†). The DFT-PBE band gap of 2H-MoS₂ indicated by the grey area is ~ 1.0 eV, which is smaller than the experimental value.^{32,55} This is owing to the well-known self-interaction error in DFT, which will underestimate the band gap. The simulated IPS ($n = 1$) and IPS ($n = 2$) are marked in red, whose features well reproduce the experimental observations. The arrowed 4-segment lines indicate 4PPE processes from the initial VB states *via* the intermediate IPSs. As our experimental observations indicate since the initial states are both located in

the VB, the intermediate IPSs could be sufficiently populated by the pump light, and thus produce the much strong 4PPE signals by the probe light.

Based on DFT calculations, the binding energies for IPS ($n = 1$) and IPS ($n = 2$) are 0.57 and 0.02 eV, respectively. It is lower than the experimental value because the DFT calculations cannot reproduce the long-range Coulomb potential correctly.¹⁹ However, the DFT calculations successfully reproduce the NFE character of the IPSs. The fitted effective masses of these two IPSs are close to 1.0 as shown in Table 1.

Fig. 3 shows an MR-*m*PPE image with a wide k_{\parallel} range for the resonance between the valence band maximum (VBM) and IPS. It is noticed that the fitting lines of the data for VBM and IPS ($n = 1$) across at the photon energy of about 1.72 eV, as shown in Fig. 2(b), indicate the existence of resonance between the states. Actually, such resonance should also be k_{\parallel} -momentum-dependent, as schematically shown in Fig. 3(a). Due to the upward dispersion of the IPSs and the downward dispersion of the VBM, the resonance may also exist between different m th excitations. As shown in Fig. 3(b), the relatively strong signals show resonance between the IPS ($n = 1$) and the VBM in the 4PPE processes at $k_{\parallel} = 0$. Moreover, as indicated by the yellow arrows, the resonance may appear between 4th IPS ($n = 1$) and 5th VBM at larger k_{\parallel} -momenta. More related MR-*m*PPE images using 1.68 and 1.63 eV excitation photon energy are also shown in Fig. 3(c) and (d), respectively. We expect that such enhanced signals at the band crossing could be a general property of IP states in the spectroscopy of semiconductors.

To investigate the role of the electron transition and relaxation process between IPS and VBM, a pulsed laser of 355 nm was used as the excitation source. We image the photoelectron distributions as a function of delay time τ between pump-probe pulses and record the coherent polarization dynamics in Fig. 4(a). The interferograms for $k_{\parallel} = 0$ in Fig. 4(a) and photoelectron distribution for $\tau = 0$ in Fig. 4(b) represent cross-sections through the three-dimensional (E, k, τ) data.

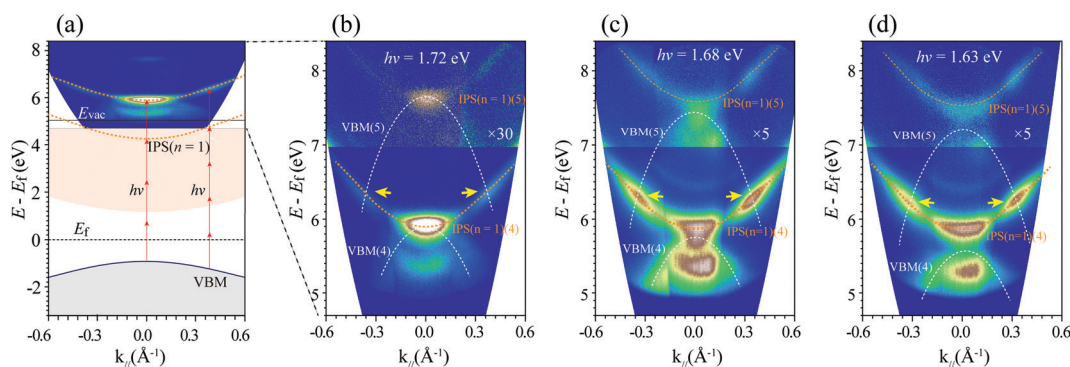


Fig. 3 (a) Schematic drawing showing the existence of resonance between the VBM and IPS ($n = 1$) in the final states. The experimentally measured $E(k_{\parallel})$ -resolved *m*PPE spectra are overlaid in the top panel. A k_{\parallel} -momentum-dependent processes of 4PPE ($k_{\parallel} = 0$) and 5PPE ($k_{\parallel} \approx 0.40 \text{ \AA}^{-1}$) are indicated by the arrowed red lines. (b) $E(k_{\parallel})$ -dependent MR-*m*PPE spectra showing the resonance between the replicas of the VBM and IPS ($n = 1$) in the 4th and 5th order excitations, marked by the bracket numbers (4) and (5). The intensity of spectra in the upper part is magnified by a factor of 30. The crossing of the bands gives relatively strong signals, indicating their resonances at the band crosses. (c) MR-*m*PPE spectra with a wide k_{\parallel} range measured at the near-resonance photon energy of 1.68 eV and (d) the off-resonance photon energy of 1.63 eV, respectively. Note that in (b) a much shorter acquisition time was used due to the strong resonance signals.

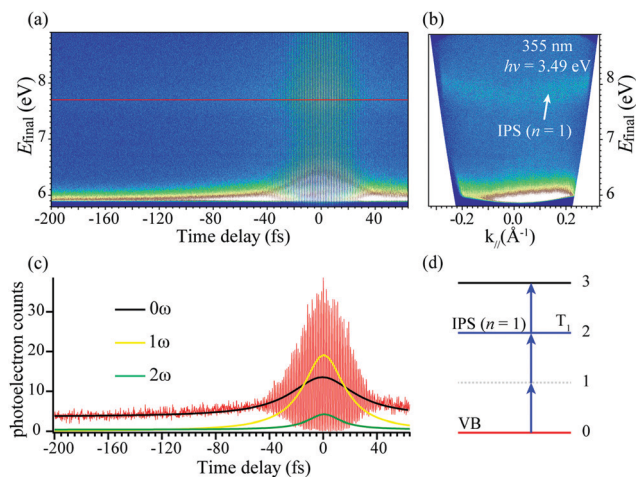


Fig. 4 3D photoemission spectra for image potential states with 3.49 eV excitation photon energy. (a) Interferogram of the photoelectron counts vs. E_{final} and scanned pump–probe time delay τ at $k_{\parallel} = 0$. (b) ARPES image for a pump–probe delay $\tau = 0$. (c) Correlation measurement was obtained by taking cross-sections through the interferogram in (a) at E_{final} for the IPS ($n = 1$). Oscillations with approximately the period of the carrier wave (~ 1.18 fs) represent interference between the pump and probe pulse-induced polarizations excited in the sample. (d) The energy diagram of the expected 3PPE processes in (b) from VB to IPS.

The distinct polarization dynamics of the IPS ($n = 1$) are evident in the correlation trace representing cross-sections through the interferogram for the E_{final} marked by the red line cutting in Fig. 4(a).

In order to show more clearly the coherence dynamics in the 3PP process, the interferogram shown in Fig. 4(c) is Fourier transformed with respect to time to generate the components at dc (0ω), the first (1ω), and the second (2ω) harmonics of the driving field. The 0ω component has contributions from the incoherent hot electron population dynamics, whereas the 1ω and 2ω components characterize the linear and second-order nonlinear coherences.⁴¹ From the cross-section of IPS at $E_{\text{final}} = 7.7$ eV shown in Fig. 4(a), the obtained interferometric two-pulse correlation trace of Fig. 4(c) is modeled with an optical Bloch equation (OBE) approach^{56,57} for a four-level scheme in Fig. 4(d). The simulation gives a population lifetime of $T_1 = 33$ fs for IPS ($n = 1$), showing an ultrafast manner. This value is comparable to the pulse duration of ~ 30 fs. We adopt an estimated error of ± 5 fs for T_1 owing to the fact that the OBE is a crude model for the bulk sample.⁵⁷

The photoexcited electron on IPS can be scattered to other bulk states through electron–phonon interactions. In this case, the lifetime of IPS is normally strongly affected by its overlap with the bulk states and the electron–phonon interaction. The lifetime measured in this report is in the same magnitude compared with other metal or semiconducting surfaces, suggesting the electronic properties of IPSS on different surfaces exhibit general character. The electron–phonon interaction of IPSS are proposed to be small.²⁵ However, the diffuse character of IPS may lead to a large orbital overlap with the bulk states, resulting in a short lifetime.

The lifetime of IPS ($n = 1$) on MoS₂ is slightly shorter than that on Cu(100) (40 fs) and Ag(100) (55 fs) surfaces.⁵⁸ On Cu(100) and Ag(100) surfaces, the IPS is located at about midgap in the projected band structure and has a small overlap with bulk states, resulting in a relatively long lifetime. In contrast, the IPS of 2H-MoS₂ is located close to the unoccupied states along Γ – K , as shown in the calculated band structure in Fig. 2(c). Such energy level alignment is similar to the Cu(111) surface, where its IPS ($n = 1$) has a lifetime of about 20 fs.⁵⁹

The IPSS on MoS₂ are found to have high energy relative to the E_{F} with a short lifetime with tens of fs, which makes it difficult for IPS to be applied in charge transport. However, we propose that there might be different techniques, which can reduce the energy of IPS and elongate their lifetime. For example, the cation or anion doping may induce dipole moment perpendicular to the surface, which may reduce the energy of IPS. If 2D MoS₂ can be wrapped into a 1D nanotube, the 1D NFE state can be formed and the structure distortion may result in a dipole moment which can stabilize the 1D NFE conduction channel. It will need further experimental and theoretical investigations.

4. Conclusion

We have measured the electronic structure and ultrafast dynamics of IPSS at the 2H-MoS₂ surface using the TR-MR-*m*PPE technique. The parabolic dispersion of the observed bands, with their energy minima located at 0.77 and 0.21 eV below the E_{vac} , can be assigned to the IPS ($n = 1$) and ($n = 2$), respectively, which can be well described by the classical hydrogenic model and have been supported by our DFT calculations. The effective masses of the IPSS are close to the rest mass of the free electron, clearly presenting the NFE character. These results also suggest a good screening effect of the excited states on the MoS₂ surface. Owing to the downward and upward dispersion of VBM and IPSS, we also observed the k_{\parallel} -momentum-dependent resonances between replicas of VBM and IPS ($n = 1$) in *m*PPE spectra. The relaxation of the photoexcited electrons at the IPS ($n = 1$) shows an ultrafast manner, in a time scale of about 33 fs. We propose the existence of IPSS, as well as their NFE character and short lifetime, can provide useful information for understanding the electron dynamics in other TMD materials.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was financially supported by the National Key R&D Program of China (Grant No. 2016YFA0200603 and 2017YFA0205004), the Anhui Initiative in Quantum Information Technologies (AHY090000), the Strategic Priority Research Program of Chinese Academy of Sciences (XDB36020200), the

Fundamental Research Funds for the Central Science Advances Universities (No. WK2340000088), and the National Natural Science Foundation of China (11620101003, 11904349, 21972129).

References

- 1 K. Giesen, F. Hage, F. J. Himpsel, H. J. Riess and W. Steinmann, *Phys. Rev. Lett.*, 1985, **55**, 300–303.
- 2 M. Kutschera, M. Weinelt, M. Rohlfing and T. Fauster, *Appl. Phys. A: Mater. Sci. Process.*, 2007, **88**, 519–526.
- 3 K. Gillmeister, M. Kiel and W. Widdra, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2018, **97**, 085424.
- 4 A. D. Miller, I. Bezel, K. J. Gaffney, S. Garrett-Roe, S. H. Liu, P. Szymanski and C. B. Harris, *Science*, 2002, **297**, 1163–1166.
- 5 J. Zhao, N. Pontius, A. Winkelmann, V. Sametoglu, A. Kubo, A. G. Borisov, D. Sánchez-Portal, V. M. Silkin, E. V. Chulkov, P. M. Echenique and H. Petek, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **78**, 085419.
- 6 M. C. Galbraith, M. Marks, R. Tonner and U. Höfer, *J. Phys. Chem. Lett.*, 2014, **5**, 50–55.
- 7 J. Zhao, M. Feng, D. B. Dougherty, H. Sun and H. Petek, *ACS Nano*, 2014, **8**, 10988–10997.
- 8 G. Dutton and X. Y. Zhu, *J. Phys. Chem. B*, 2002, **106**, 5975–5981.
- 9 S. Tan, A. Argondizzo, J. Ren, L. Liu, J. Zhao and H. Petek, *Nat. Photonics*, 2017, **11**, 806–812.
- 10 M. V. Mamonova, V. V. Prudnikov and I. A. Prudnikova, *Surface physics: theoretical models and experimental methods*, CRC Press, 2016.
- 11 S. Tognolini, S. Ponzoni, F. Sedona, M. Sambri and S. Pagliara, *J. Phys. Chem. Lett.*, 2015, **6**, 3632–3638.
- 12 X. Cui, C. Wang, A. Argondizzo, S. Garrett-Roe, B. Gumhalter and H. Petek, *Nat. Phys.*, 2014, **10**, 505–509.
- 13 X. Y. Zhu, *J. Phys. Chem. B*, 2004, **108**, 8778–8793.
- 14 K. F. Mak, K. He, J. Shan and T. F. Heinz, *Nat. Nanotechnol.*, 2012, **7**, 494–498.
- 15 O. V. Yazyev and A. Kis, *Mater. Today*, 2015, **18**, 20–30.
- 16 P. M. Echenique and J. B. Pendry, *Prog. Surf. Sci.*, 1989, **32**, 111–172.
- 17 P. M. Echenique and M. E. Uranga, *Surf. Sci.*, 1991, **247**, 125–132.
- 18 U. Höfer, I. L. Shumay, C. Reuß, U. Thomann, W. Wallauer and T. Fauster, *Science*, 1997, **277**, 1480–1482.
- 19 V. M. Silkin, J. Zhao, F. Guinea, E. V. Chulkov, P. M. Echenique and H. Petek, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **80**, 121408.
- 20 N. Armbrust, J. Güdde and U. Höfer, *New J. Phys.*, 2015, **17**, 103043.
- 21 D. Gugel, D. Niesner, C. Eickhoff, S. Wagner, M. Weinelt and T. Fauster, *2D Mater.*, 2015, **2**, 045001.
- 22 M. Feng, J. Zhao and H. Petek, *Science*, 2008, **320**, 359–362.
- 23 J. Zhao, M. Feng, J. Yang and H. Petek, *ACS Nano*, 2009, **3**, 853–864.
- 24 S. Hu, J. Zhao, Y. Jin, J. Yang, H. Petek and J. G. Hou, *Nano Lett.*, 2010, **10**, 4830–4838.
- 25 J. Zhao and H. Petek, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 075412.
- 26 R. Yamamoto, T. Yamada, M. Taguchi, K. Miyakubo, H. S. Kato and T. Munakata, *Phys. Chem. Chem. Phys.*, 2012, **14**, 9601–9605.
- 27 B. Borca, C. Castenmiller, M. Tsvetanova, K. Sotthewes, A. N. Rudenko and H. J. W. Zandvliet, *2D Mater.*, 2020, **7**, 035021.
- 28 T. Cao, G. Wang, W. Han, H. Ye, C. Zhu, J. Shi, Q. Niu, P. Tan, E. Wang, B. Liu and J. Feng, *Nat. Commun.*, 2012, **3**, 887.
- 29 B. Radisavljevic and A. Kis, *Nat. Mater.*, 2013, **12**, 815–820.
- 30 J. Madeo, M. K. L. Man, C. Sahoo, M. Campbell, V. Pareek, E. L. Wong, A. Al-Mahboob, N. S. Chan, A. Karmakar, B. M. K. Mariserla, X. Li, T. F. Heinz, T. Cao and K. M. Dani, *Science*, 2020, **370**, 1199–1203.
- 31 A. Tanaka, N. J. Watkins and Y. Gao, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **67**, 113315.
- 32 S. K. Mahatha, K. D. Patel and K. S. R. Menon, *J. Phys.: Condens. Matter*, 2012, **24**, 475504.
- 33 P. Hein, A. Stange, K. Hanff, L. X. Yang, G. Rohde, K. Rossnagel and M. Bauer, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2016, **94**, 205406.
- 34 R. Wallauer, J. Reimann, N. Armbrust, J. Güdde and U. Höfer, *Appl. Phys. Lett.*, 2016, **109**, 162102.
- 35 R. Wallauer, P. Maruhn, J. Reimann, S. Zoerb, F. Kraus, J. Güdde, M. Rohlfing and U. Höfer, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2020, **102**, 125417.
- 36 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, *Phys. Rev. Lett.*, 2010, **105**, 136805.
- 37 M. Palummo, M. Bernardi and J. C. Grossman, *Nano Lett.*, 2015, **15**, 2794–2800.
- 38 F. Liu, M. E. Ziffer, K. R. Hansen, J. Wang and X. Zhu, *Phys. Rev. Lett.*, 2019, **122**, 246803.
- 39 M. Reutzler, A. D. Li and H. Petek, *Phys. Rev. X*, 2019, **9**, 011044.
- 40 H. Petek and S. Ogawa, *Prog. Surf. Sci.*, 1997, **56**, 239–310.
- 41 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 42 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1997, **78**, 1396.
- 43 S. Grimme, J. Antony, S. Ehrlich and H. Krieg, *J. Chem. Phys.*, 2010, **132**, 154104.
- 44 R. Addou, L. Colombo and R. M. Wallace, *ACS Appl. Mater. Interfaces*, 2015, **7**, 11921–11929.
- 45 H. Xu, D. Han, Y. Bao, F. Cheng, Z. Ding, S. J. R. Tan and K. P. Loh, *Nano Lett.*, 2018, **18**, 5085–5090.
- 46 J. Lin, J. Zhong, S. Zhong, H. Li, H. Zhang and W. Chen, *Appl. Phys. Lett.*, 2013, **103**, 063109.
- 47 S. Tognolini, S. Achilli, L. Longetti, E. Fava, C. Mariani, M. I. Trioni and S. Pagliara, *Phys. Rev. Lett.*, 2015, **115**, 046801.
- 48 M. Reutzler, A. Li and H. Petek, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2020, **101**, 011044.
- 49 D. B. Dougherty, P. Maksymovych, J. Lee, M. Feng, H. Petek and J. T. Yates, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 3398–3407.
- 50 M. Muntwiler and X. Y. Zhu, *New J. Phys.*, 2008, **10**, 113018.
- 51 Y. Li, A. Chernikov, X. Zhang, A. Rigosi, H. M. Hill, A. M. van der Zande, D. A. Chenet, E.-M. Shih, J. Hone and T. F. Heinz, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 205422.

- 52 A. Molina-Sánchez and L. Wirtz, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 155413.
- 53 T. Cheiwchanchamnangij and W. R. L. Lambrecht, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 205302.
- 54 E. V. Chulkov, V. M. Silkin and P. M. Echenique, *Surf. Sci.*, 1999, **437**, 330–352.
- 55 M. M. Ugeda, A. J. Bradley, S.-F. Shi, F. H. da Jornada, Y. Zhang, D. Y. Qiu, W. Ruan, S.-K. Mo, Z. Hussain, Z.-X. Shen, F. Wang, S. G. Louie and M. F. Crommie, *Nat. Mater.*, 2014, **13**, 1091–1095.
- 56 W. Nessler, S. Ogawa, H. Nagano, H. Petek, J. Shimoyama, Y. Nakayama and K. Kishio, *Phys. Rev. Lett.*, 1998, **81**, 4480–4483.
- 57 S. Tan, Y. Dai, S. Zhang, L. Liu, J. Zhao and H. Petek, *Phys. Rev. Lett.*, 2018, **120**, 126801.
- 58 I. L. Shumay, U. Höfer, C. Reuß, U. Thomann, W. Wallauer and T. Fauster, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **58**, 13974–13981.
- 59 K. Schubert, A. Damm, S. V. Eremeev, M. Marks, M. Shibuta, W. Berthold, J. Güdde, A. G. Borisov, S. S. Tsirkin, E. V. Chulkov and U. Höfer, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 205431.