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## Real-time monitoring of trap dynamics reveals the electronic states that limit charge transport in crystalline organic semiconductors†

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Organic semiconductors (OSCs) have the potential to become ubiquitous in our lives as part of various optoelectronic devices given their low-cost processing, light weight, and the opportunities that they offer for designing new materials with “on demand” properties. Many OSCs have achieved remarkable performances, however, a complete understanding of the fundamental processes that govern their properties is still lacking. One such process is charge carrier trapping, a phenomenon that profoundly impacts the performance and stability of devices. Here we report on a comprehensive study on the physics of traps resulting from microstructure changes by monitoring the generation/annihilation of traps in 2,8-difluoro-5,11-bis(triethylsilylethynyl)anthradithiophene (diF-TES ADT), a high-mobility small molecule OSC. The variations in charge carrier mobility measured in organic field-effect transistors (OFETs) deliberately exposed to solvent vapor are correlated with the density and energetic distribution of the electronic states in the band gap of the OSC in all intermediate states of the microstructural transformations. Given the strong dependence of the electrical properties on the film microstructure, such information is instrumental in identifying performance-limiting processes in devices and subsequently guiding material processing to achieve intrinsic limits. The discovery of defect-tolerant intermediate crystalline motifs may provide new pathways for fabricating stable, high-performance devices for next-generation low-cost electronics.

## Introduction

Organic semiconductors (OSCs) are gaining attention for incorporation in the next-generation flexible, bendable and stretchable optoelectronic devices. Key attributes that make them attractive

### New concepts

Organic semiconductors can become ubiquitous in optoelectronics given their low-cost processing and compatibility with flexible substrates – properties arising from their weak intermolecular interactions. Unfortunately, these interactions also make them susceptible to defect formation: localized electronic states can trap charge carriers, a phenomenon that profoundly impacts the performance and stability of devices. Here, we probe for the first time the transformations in the density and energetic landscape of electronic traps during microstructure evolution in organic semiconductor films. Monitoring in real-time the trap density of states gave us unprecedented access to the complex structural dynamics occurring in high-mobility materials and the underpinning structure-property relationships. Given the strong dependence of electrical properties on the film microstructure, such measurements are instrumental in identifying performance-limiting processes in optoelectronic devices and subsequently guiding material processing to achieve intrinsic limits. Our results simultaneously provide the electrical properties and the energetic distribution of traps of all intermediate states during the microstructural transformations. Access to such information brings an important new perspective to understanding the fundamental processes governing charge transport in organic semiconductors and may result in the discovery of defect-tolerant intermediate crystalline motifs that can be selectively frozen in to achieve efficient devices.

are their low-cost processing, lightweight, tunability of properties, and compatibility with flexible substrates.<sup>1–5</sup> These properties are a direct consequence of the weak intermolecular interactions inherent in these materials, but unfortunately these interactions also make OSCs susceptible to defect formation. Localized electronic states introduced in the band gap can trap charge carriers, a phenomenon that has a profound impact on the mechanism of charge transport, as well as on the performance and stability of optoelectronic devices.<sup>6–14</sup> Hence, their detection and characterization is crucial to understanding performance-limiting processes, and subsequently guiding the processing and handling that is necessary for engineering high-performance optoelectronic devices.<sup>1,6,9,15–19</sup> Several experimental tools are available for this purpose, including optical, thermal and electrical methods that

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provide access to the density, spatial and energetic distribution of traps.<sup>12,20–23</sup> In particular, the energetic distribution of traps, also referred to as the trap DOS spectrum, provides information about the distribution of electronic states as a function of energy within the band gap of the OSC.<sup>22,24–27</sup> Access to this information has yielded improvements in device performance and reliability. For example, new material design rules have been proposed to enable trap-free charge transport in OSCs.<sup>28</sup> The nature of device interfaces has been re-evaluated in order to reduce trap formation during operation of organic field-effect transistors (OFETs),<sup>29</sup> and molecular additives have been incorporated into polymer films to enhance the electrical stability of organic diodes.<sup>8</sup> Systematic studies where traps are deliberately introduced into the OSCs to examine their effect on the DOS spectrum and device performance are very insightful as they can reveal the precise nature and origin of traps, provide a quantitative understanding of their effects, and therefore guide device design and film processing by pinpointing the phenomena that could lead to more trap-tolerant systems. However, only a few such studies have been reported. For example, the effect of O<sub>2</sub> exposure,<sup>30</sup> X-ray irradiation,<sup>8</sup> proton and He<sup>+</sup> irradiation,<sup>31</sup> on the DOS spectrum of OSCs have been studied.

Here we monitored in real time the generation and annihilation of traps in OSCs in response to microstructural changes and percolation of impurity centres, and examined the resulting impact on the performance of OFETs. The devices were deliberately exposed to solvent vapor and we probed the density and energetic landscape of electronic traps of all intermediate states during the microstructural transformation. The measurements provided access to the dynamics of trap formation and

healing as a result of the complex processes taking place within the OSC upon interaction with the solvent. We focused on 2,8-difluoro-5,11-bis(triethylsilyl)ethynylanthradithiophene (diF-TES ADT), a small molecule OSC that has demonstrated remarkable electrical properties, with mobilities greater than 10 cm<sup>2</sup> V<sup>−1</sup> s<sup>−1</sup> in optimized devices with near ideal current-voltage characteristics.<sup>32,33</sup> diF-TES ADT has two isomers: *syn* and *anti* (see molecular structures in Fig. 1); the material typically consists of a 40–60 isomeric mixture. The coexistence of isomers in the transistor channel introduces a peak in the DOS spectrum at *ca.* 0.3–0.4 eV from the valence band edge, corresponding to deep traps responsible for the reduced device performance when compared to isomerically pure samples, where the peak is absent.<sup>34</sup> The isomers can be separated through elaborate synthetic procedures,<sup>35,36</sup> or meticulous processing.<sup>34</sup> Recently, polarized Micro Raman spectroscopy measurements indicated that the *syn* and *anti* isomers phase-segregate into pure-isomer crystalline grains that coexist intimately in the transistor channel of mix-isomer films.<sup>37</sup> Hence, the devices can be of two types, depending on the distribution of grains across the transistor channel. The first type consists of a mixture of *syn* and *anti* crystalline grains, as illustrated in Fig. 1a, and the second contains only grains of one isomer type in the channel: either only *syn* (top panel of Fig. 1b) or only *anti* (bottom panel of Fig. 1b). Consequently, the mix-devices exhibit a peak in the trap DOS spectrum, while the pure isomer devices do not, and hence we will use the presence/absence of the peak as a way to distinguish between the two device types in order to individually monitor the response of these devices when exposed to a solvent vapor. We discovered that exposing an



**Fig. 1** diF-TES ADT OFET device structure and film modifications during exposure to ethanol vapor. (a) Device consisting of a mixture of *syn* and *anti* isomer crystalline grains in the transistor channel between the source (S) and drain (D) electrodes. (b) A phase separation occurs upon exposure to ethanol vapor and each isomer re-crystallizes into grains consisting of pure *syn* (top) or pure *anti* (bottom) isomers. Shades of orange and red are used to symbolize different orientations of individual grains of *syn* and *anti* isomers, respectively. (c) The molecular structures of *syn* (top) and *anti* (bottom) isomers of diF-TES ADT.

OFET containing a film of isomer mixture to ethanol vapor induces a phase separation of the two isomers *via* reorganization of molecules in the transistor channel, which is detected as a gradual reduction in intensity, followed by the disappearance of the peak in the DOS spectrum. In addition, a reduction in the overall density of states was also evident, resulting from microstructural rearrangement of the film in the process of reaching thermodynamic equilibrium. Correspondingly, a 30% increase in charge carrier mobility was observed during the time of exposure. On the other hand, in devices with pure isomer crystallites, exposure to the solvent vapor generated band gap states manifested in the DOS spectrum as a peak centered at *ca.* 0.22 eV from the valence band edge, likely originating from impurity centers percolating into the film. The generation of these states results in a decrease in the device mobility by 20%.

## Experimental

OFETs were fabricated in the bottom-gate, bottom-contact geometry with a heavily doped n-type Si wafer and a 200 nm thermally grown SiO<sub>2</sub> layer serving as the gate and the gate dielectric, respectively. A 3 nm Ti layer followed by a 40 nm Au layer was evaporated through a shadow mask to define the source and the drain electrodes. The prefabricated substrates were treated with pentafluorobenzenethiol (PFBT) self-assembled monolayer to improve charge injection and film microstructure.<sup>32</sup> A 1.5 wt% solution of a 40/60 diF-TES ADT isomeric mixture in room-temperature chlorobenzene was then spin coated onto the prefabricated substrate according to the procedure reported elsewhere.<sup>32</sup> The resulting film thickness was  $\sim 60$  nm. Electrical measurements were performed using an Agilent 4155 C parameter analyzer under an inert N<sub>2</sub> or N<sub>2</sub>/ethanol vapor (for measurements obtained during ethanol exposure) environment. Room-temperature anhydrous (histological grade) ethanol from Fisher scientific Inc. was used as purchased, with no further treatment. The exposure of OFETs to ethanol vapor was performed in a nitrogen atmosphere and in dark.

In order to access the trap DOS spectrum, we employed Grünewald's model, which uses the linear-regime transfer characteristics of an OFET (drain current  $I_D$  as a function of gate-source voltage  $V_{GS}$  at a low, fixed drain-source voltage  $V_{DS}$ ) to extract the trap DOS as a function of energy from the valence band edge.<sup>38–40</sup> Here, we briefly outline the essential steps necessary to perform the analysis. First, the interface potential at the OSC/dielectric interface ( $V_0$ ) is obtained as a function of gate-source voltage above flat-band ( $U_{GS} = V_{GS} - V_{FB}$ ) by numerically solving the following equation:

$$\exp\left(\frac{eV_0}{kT}\right) - \frac{eV_0}{kT} - 1 = \frac{e}{kT} \frac{\epsilon_i d}{\epsilon_s l \sigma_0} \times \left[ U_{GS} \sigma(U_{GS}) - \int_0^{U_{GS}} \sigma(\widetilde{U}_{GS}) d\widetilde{U}_{GS} \right] \quad (1)$$

where  $\epsilon_i$  and  $l$  are the relative permittivity and the thickness of the dielectric, respectively,  $\epsilon_s$  and  $d$  are the relative permittivity and the

thickness of the semiconductor, respectively,  $\sigma(U_{GS})$  is the field-effect conductivity (evaluated from the linear regime transfer characteristics) and  $\sigma_0$  is the conductivity at flat band voltage, assumed to be the turn-on voltage ( $V_{ON}$ ) of the device. The function  $V_0(U_{GS})$  is used to determine the total whole density from,

$$p(V_0) = \frac{\epsilon_0 \epsilon_i^2}{\epsilon_s l^2 e} U_{GS} \left( \frac{dV_0}{dU_{GS}} \right)^{-1} \quad (2)$$

The trap DOS is then obtained by numerically differentiating the total hole density with respect to  $V_0$ , *i.e.*,

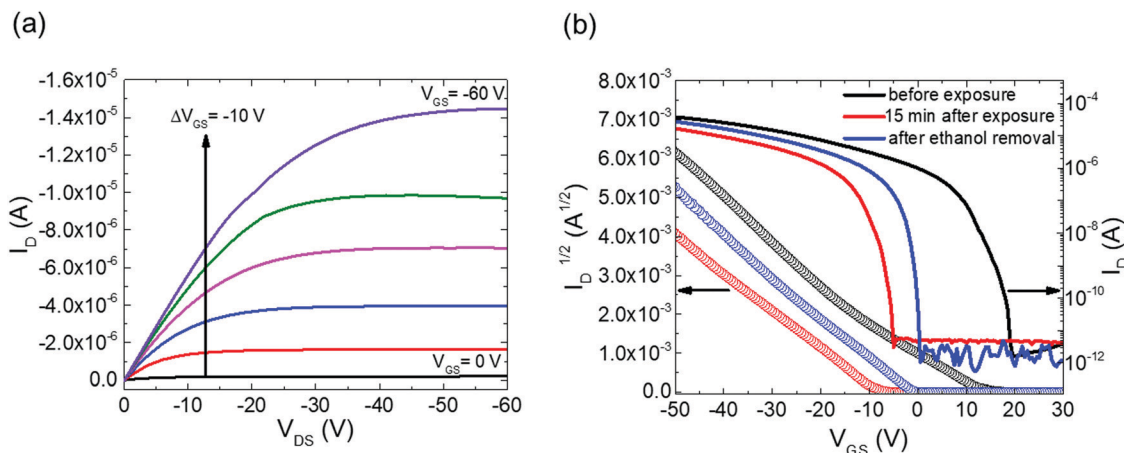
$$N(E) \approx \frac{1}{e} \frac{dp(V_0)}{dV_0} \quad (3)$$

where  $E = eV_0$  is the energy of the trapping state w.r.t the Fermi level. diF-TES ADT yields p-channel OFETs and hence in these measurements we access electronic states above the valence band edge. The result is the trap DOS spectrum as a function of energy in the bandgap of the OSC.

## Results and discussion

Fig. 2a shows the output characteristics of a device consisting of mixed isomers ( $I_D$  vs.  $V_{DS}$  at various fixed  $V_{GS}$ ) obtained before exposure to the solvent vapor. These characteristics are near-ideal, as indicated by the linearity of the curves at low  $V_{DS}$  and clear saturation attained at high  $V_{DS}$ . The absence of S-shaped current-voltage curves close to  $V_{DS} = 0$  is an indication of low contact resistance.<sup>32,41</sup> In Fig. 2b we include the transfer characteristics in the saturation regime ( $I_D$  vs.  $V_{GS}$  at  $V_{GS} = -60$  V) before exposure to ethanol vapor (black), after 15 min into exposure (red), and immediately after the ethanol container was removed (blue). Clearly, the exposure to solvent vapor is inducing a significant change in the current-voltage characteristics. Intrigued by these experimental observations, we monitored *in situ* how the trap DOS spectrum evolves during exposure to the solvent vapor. To perform this analysis, the devices were first measured under N<sub>2</sub> and then an open jar (diameter  $\sim 1.5$  cm) of ethanol was transferred into the measurement chamber and the linear-regime transfer characteristics were recorded immediately, at regular time intervals of 15 s, for a duration of 15 min. The DOS spectra were evaluated following the procedure described above.

In Fig. 3a, we plot the time evolution of the trap DOS spectrum of a device consisting of mixed isomers when exposed to ethanol vapor. The function  $V_0(U_{GS})$  is included in Fig. S1 in ESI.† Before exposure, *i.e.*, at  $t = 0$  s (red curve), the spectrum resembles an exponential DOS close to the valence band edge, with a peak at 0.36 eV, reminiscent of a device with mixed isomer crystalline grains in the channel (Fig. 1a). While exponential DOS are generally associated with disorder in the film originating from thermal molecular motions and structural defects, a peak in the DOS spectrum is due to discrete electronic states in the bandgap, originating from chemical impurities or a predominant defect.<sup>6</sup> In our measurements, the peak arises due to deep traps originating from the presence of one



**Fig. 2** Current–voltage characteristics of a diF-TES ADT OFET device exposed to solvent vapor. (a)  $I_D$  vs.  $V_{DS}$  plot at various fixed  $V_{GS}$  (0 V to  $-60$  V in  $-10$  V increments) measured before exposure to the ethanol vapor. (b)  $I_D$  vs.  $V_{GS}$  plot at  $V_{DS} = -60$  V obtained before exposure to ethanol vapor (black), after 15 min of exposure (red) and after removal of ethanol from the measurement chamber (blue). The left and right axes show the square root and logarithm of the drain current, respectively.

isomer in the host crystal of the other, details on the mechanism of trapping were discussed elsewhere.<sup>34</sup> After 15 s of exposure (orange curve), the amplitude of the peak diminishes and the peak is completely annihilated at  $t = 30$  s. We attribute this transformation to a gradual isomer separation, facilitated by a modification of the film microstructure in the presence of ethanol vapor. Ethanol molecules interact with the OSC film, promoting molecular rearrangement within the layer to reach a more energetically favorable state.<sup>42</sup> The *syn* isomer is more polar in nature than the *anti* isomer and therefore it is possible that the polar ethanol molecules selectively extract the *syn* isomers from the film facilitating the isomer separation. While the phase separation is evident, the type of isomer accessed in our measurements is not clear due to the complex nature of this separation. Considering the vertical segregation, the polar  $\text{SiO}_2$  substrate would favour the assembly of the more polar isomer (*syn*) on its surface,<sup>34</sup> however, the denser crystalline lattice of the *anti*-isomer would favour its accumulation on the bottom.<sup>35</sup> A horizontal phase separation also occurs, leading to the formation of pure isomer crystalline sub-domains.<sup>37</sup> Not surprisingly, but also confirmed by density functional theory (DFT) calculations, the lowest energy corresponds to isomer demixing,<sup>34</sup> hence the film progresses towards isomerically pure crystallites (Fig. 1b) during solvent exposure. In fact, when measuring an array of OFET devices fabricated from a *syn/anti* mixture, we typically find that around 85% of devices consist only of one isomer, as indicated by the absence of a peak in the DOS spectrum, which is in agreement with the energy considerations. Upon further exposure to the solvent vapor, a reduction in the overall trap densities is also evident in the DOS spectrum (see the curves from orange to violet in Fig. 3a). Solvent-vapor annealing is known to cause microstructural rearrangement, allowing the film initially trapped kinetically in non-equilibrium states to attain thermodynamic equilibrium.<sup>43,44</sup> A decrease in the overall trap densities is therefore attributed to molecular reorganization in the transistor channel to minimize the total potential energy until the system reaches thermodynamic equilibrium.

Thus, these measurements, where we were able to monitor the reduction in the peak intensity followed by its disappearance, and then the minimization of the overall trap densities, gave us real-time insights into the dynamics of microstructural transformations occurring in the film containing mixed isomer crystalline grains. We discovered that reorganization of molecules in the transistor channel in the presence of solvent vapor is a two-step process: first, isomer separation occurs to minimize the total energy of the system, followed by a reduction in the crystalline defect density to reach the state of thermodynamic equilibrium. Structural rearrangement occurs within a minute of solvent exposure, similar to the case of TES ADT films.<sup>44</sup>

To obtain deeper insights into the trap parameters and how they alter after solvent exposure, we modeled the DOS spectra at  $t = 0$  s and  $t = 900$  s using two exponential functions to describe the distribution of shallow and deep states, respectively, with the addition of a Gaussian distribution to model the spectrum at  $t = 0$  s according to the following equation:

$$N(E) = N_1 \exp\left(-\frac{E}{E_1}\right) + N_2 \exp\left(-\frac{E}{E_2}\right) + A \exp\left[-\frac{(E - E_{\text{peak}})^2}{2\sigma^2}\right] \quad (4)$$

where  $E_1$ ,  $E_2$  are the characteristic decay energies and  $N_1$ ,  $N_2$  are the amplitudes of the respective exponential distributions,  $E_{\text{peak}}$  defines the position of the Gaussian distribution with an amplitude  $A$  and standard deviation  $\sigma$ . The model parameters and fits to experimental data are summarized in Table 1 and Fig. S2 in ESI,<sup>†</sup> respectively. While the characteristic energy for the double exponential distributions remains the same over the course of ethanol exposure ( $\sim 21$  meV for shallow states and  $\sim 86$  meV for deeper states), the density of shallow and deep states decreases by 65% and 12% respectively after 15 min of solvent exposure. The Gaussian distribution centered at 0.36 eV from the valence band edge represents the peak in DOS spectrum at  $t = 0$  s.





**Fig. 3** (a) Evolution of the trap DOS spectrum of a device with mixed isomer crystallites. The DOS curves are plotted as a function of energy from the valence band edge and clearly illustrate the annihilation of the peak at 0.36 eV. The DOS was evaluated every 15 s during exposure to ethanol vapor and the time evolution of the spectrum from  $t = 0$  s (red) to  $t = 900$  s (violet) is shown. Inset includes a subset of the trap DOS spectra offset for clarity. (b) Time evolution of the charge carrier mobility during the period of solvent exposure. The value is normalized to the initial value obtained before exposure to ethanol.

In order to understand how the process of trap reduction influences the performance of devices, we monitored the evolution of the charge carrier mobility ( $\mu$ ), during the entire time of exposure to ethanol. The mobility was extracted from the slope of  $I_D$  vs.  $V_{GS}$  curve using the following equation,<sup>45</sup>

$$\mu = \frac{L}{C_i W V_{DS}} \frac{\partial I_D}{\partial V_{GS}} \quad (5)$$

where  $L$  and  $W$  are the length and width of the transistor channel,  $C_i = 17.3 \text{ nF cm}^{-2}$  is the areal capacitance of the 200 nm thick  $\text{SiO}_2$  dielectric. Mobilities of devices were in the order of  $0.1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , on par with values reported on this material in OFETs with  $\text{SiO}_2$  dielectric and in similar device architectures.<sup>34,35</sup> Higher mobilities are achievable with Cytop dielectric in the top-gate bottom-contact geometry due to reduced interfacial trap densities at the dielectric/semiconductor interface and lower contact resistance facilitated by the staggered device geometry.<sup>29,35</sup> Our choice of device type and geometry was motivated by the need to increase the sensitivity of trap detection by increasing the overall trap densities. Additionally, this architecture allows the film to better reorganize during

solvent exposure without being constrained by the conformal dielectric and gate electrode on the top. Fig. 3b shows the time evolution of the device mobility normalized to the initial value (at  $t = 0$  s). Three different regimes can be identified. In the first regime (shaded red region), a drastic increase in mobility is observed, which corresponds to the annihilation of the peak in the DOS spectrum resulting from solvent-induced isomer separation. The schematic in Fig. 4 illustrates different snapshots of the energetic distribution of traps in the band gap over the course of ethanol exposure. Panel a corresponds to  $t = 0$  s, where deep states exist in the band gap, indicated in red, in addition to the shallow tail states shown in black. The density of deep states is reduced at  $t = 15$  s (Fig. 4b) and they completely vanish at  $t = 30$  s (Fig. 4c), as indicated by the disappearance of the peak in the DOS spectrum. In the second region of Fig. 3b, enclosed by yellow shading, the mobility is increasing at a much slower rate, a phenomenon which is attributed to molecular reorganization in the transistor channel to minimize the total potential energy, which translates into a corresponding reduction in the overall trap densities evident in the DOS spectrum. This reduction is reflected in Fig. 4d which depicts a decrease in the number of shallow tails states after exposure to the solvent vapor. Finally, in the third region in Fig. 3b, bounded by blue shading (*i.e.*, after about 750 s), the mobility appears to have plateaued, indicating molecular assembly in the lowest potential energy configuration and that the system has reached thermodynamic equilibrium. In a previous study we have showed that such a stable configuration can be achieved by applying gentle vibrations during crystallization of the film.<sup>34</sup>

Since the above measurements were taken over short periods of time (15 s), it is likely that the device was under excessive bias stress. Bias stress effects can also induce charge carrier traps.<sup>6,46</sup> Hence, there are, in fact, two possible contributions to the changes in the trap distribution observed above; bias stress

**Table 1** Model parameters used for the double exponential and Gaussian distributions employed to model the DOS spectra of a device with mixed isomers. The spectrum at  $t = 900$  s was modelled only using a double exponential distribution and therefore the Gaussian parameters are not applicable

| Model parameter                            | $t = 0$ s            | $t = 900$ s          |
|--|----------------------|----------------------|
| $N_1$ ( $\text{eV}^{-1} \text{ cm}^{-3}$ ) | $4.9 \times 10^{22}$ | $1.7 \times 10^{22}$ |
| $E_1$ (meV)                                | 21.2                 | 20.6                 |
| $N_2$ ( $\text{eV}^{-1} \text{ cm}^{-3}$ ) | $2.5 \times 10^{20}$ | $2.2 \times 10^{20}$ |
| $E_2$ (meV)                                | 85.5                 | 85.5                 |
| $A$ ( $\text{eV}^{-1} \text{ cm}^{-3}$ )   | $1.0 \times 10^{19}$ | N/A                  |
| $E_{\text{peak}}$ (eV)                     | 0.36                 | N/A                  |
| $\sigma$ (meV)                             | 29.0                 | N/A                  |



**Fig. 4** Snapshots of the energetic distribution of localized band gap states at different times during microstructure reorganization in a device with mixed isomer grains: (a)  $t = 0$  s, (b)  $t = 15$  s, (c)  $t = 30$  s and (d)  $t = 900$  s. Shallow and deep states in the band gap are represented in black and red lines, respectively.

and microstructure reorganization in the presence of ethanol vapor. In order to clarify the individual contribution of each, we performed several control experiments. First, characterization of devices based on mixed isomers was carried out in  $N_2$  under the same time intervals, but in the absence of ethanol, to eliminate the effect caused by the solvent vapor, results are shown in Fig. S3 in ESI.† Contrarily to the measurements performed under ethanol exposure, the peak in the DOS spectra persists, which confirms that bias stress is not responsible for the observed changes. In the second test experiment, the devices were exposed to ethanol vapor, but measurements were taken only before exposure and after 15 minutes, thus, ensuring that the devices were not stressed during the exposure;

results are displayed in Fig. S4 in ESI.† The annihilation of the peak after 15 min of exposure in the absence of any bias stress unequivocally confirms that the changes observed in the DOS spectrum are solely caused by the effect of the solvent vapor. Also shown in the same plot is the DOS spectrum evaluated after the solvent was removed from the chamber following exposure (blue curve). The peak does not reappear after the jar of ethanol is removed, implying that once a device with mixed isomers transforms into the more stable device consisting only of one isomer, it remains in that state.

Next, we tested the response of devices consisting of pure isomer crystallites during exposure to ethanol vapor. Measurements were taken both on OFETs where the isomer separation



**Fig. 5** (a) Evolution of the trap DOS spectrum of a device consisting of pure isomers. The DOS curves are plotted as a function of energy from the valence band edge and clearly illustrate the generation of a peak at 0.22 eV. The DOS was evaluated every 30 s during exposure to ethanol vapor and the time evolution of the spectrum from  $t = 0$  s (red) to  $t = 900$  s (violet) is shown. Inset shows a subset of the trap DOS spectra offset for clarity. (b) Time evolution of the charge carrier mobility during the period of solvent exposure. The value is normalized to the initial value obtained before exposure to ethanol.

took place during sample fabrication, as well as on solvent-annealed samples discussed above; the results were similar. Fig. 5a shows the DOS spectrum evaluated every 30 s over a course of 15 min. The function  $V_0(U_{GS})$  is included in Fig. S5 of ESI.† Initially, the DOS spectra represent an exponential DOS with the absence of a peak (red curve). A snapshot of the trap distribution corresponding to this scenario is provided in Fig. S6a of ESI.† After  $t = 30$  s, a progressive generation of a peak centered at 0.22 eV from the valence band edge is evident in the spectra. The establishment of the peak is initiated with a broadening of the DOS at  $t = 30$  s, followed by a gradual increase in peak amplitude from  $t = 60$  s to  $t = 900$  s. The normalized mobility *vs.* time plot for the same device (Fig. 5b) shows a monotonic decrease in the mobility during the entire duration of exposure (an overall  $\sim 20\%$  decrease), correlating with the increasing amplitude of the peak observed in the DOS spectrum. Additional electronic states are generated in the band gap during the period of exposure to the solvent vapor (see Fig. S6b of ESI.†), which were absent initially, explaining the creation of the peak in the DOS spectrum and therefore the decrease in mobility. Establishing the nature of the trap giving rise to the peak is challenging since this appears to be a metastable state; the samples return to the initial “no-peak” state in a matter of seconds as soon as the ethanol vapor is removed. We do not exclude that a reverse modification of the film microstructure, where the molecules reorganize to form mixed isomer crystallites, occurs. Nevertheless, it should be noted that the trap depth of the electronic states generated is different from that in the earlier case of 0.36 eV. Additionally, since this device type is already in its lowest energy configuration, which corresponds to a state where the isomers are separated, the energy penalty necessary to convert it to the less stable configuration of a device consisting of mixed isomer grains would be significant. Hence, this scenario is highly unlikely, and the peak probably originated from a different source. One such likelihood is that ethanol (or the impurities present in it), upon percolation into the film, acts as an impurity center giving rise to discrete electronic states in the band gap, with depths that are dependent on the electric field induced by the dipole moment of the polar ethanol molecules.<sup>47</sup> Trap parameters extracted by modeling the DOS spectra at  $t = 0$  s and  $t = 900$  s using eqn (4) are summarized in Table S1 in ESI.† and the fits to experimental spectra are provided in Fig. S7 of ESI.† The densities of both shallow and deep states do not change significantly over the course of ethanol exposure, however, the increase in the characteristic decay energy of both shallow and deep states indicates a broadening of the DOS. Such a broadening can take place when the crystal lattice is distorted due to the presence of impurity centers which introduces the peak in the DOS spectrum represented by the Gaussian centered at 0.22 eV. With new insights gained on the impact of impurity centres on the DOS, we took a closer look at the trap DOS spectrum of the mixed isomer device (Fig. 3a); it appears that a new peak centered around 0.2 eV begins to form after prolonged ethanol exposure (*i.e.*, blue/purple curves at  $t > 360$  s), most likely due to the metastable impurity centers percolating into the film following

isomer separation. Comparing Fig. 3a and 5a, the following can be inferred about the dynamics of trap annihilation *vs.* trap formation; the annihilation of traps due to isomer separation in mixed isomer films is a fast process and takes place within the first minute of exposure to ethanol vapor, while the formation of traps due to impurity centers occurs after about 5 min of ethanol exposure.

Control experiments similar to those performed on mixed isomer devices were also carried out on this device type in order to confirm that the generation of the peak in DOS spectrum is solely induced by the interaction of the film with ethanol vapor and not a bias stress effect. The DOS spectra in Fig. S8 of ESI.† evaluated over the course of bias stress measurements show no evidence for the establishment of a peak. Similarly, the DOS spectra of a device with pure isomers shows the generation of a peak after 15 min of exposure in the absence of bias stress (Fig. S9, ESI.†), confirming that such a peak was indeed caused by the effect of ethanol vapor. The absence of the peak in the DOS spectrum evaluated after the removal of the ethanol vapor (blue curve) suggests that such discrete trapping states are evident only in the presence of the solvent vapor and supports our earlier hypothesis of ethanol acting as a metastable impurity center.

## Conclusions

In summary, we accessed in real time the generation and annihilation of charge carrier traps in a small molecule OSC and evaluated the resulting effect on charge transport. The traps originated from microstructure modifications or introduction of impurity centers upon exposure to a solvent vapor. A sharp increase/decrease in the charge carrier mobility of the devices exposed to ethanol vapor coincides with the elimination/formation of a predominant trap, as suggested by the gradual annihilation/generation of a peak in the DOS spectrum. A milder enhancement in mobility observed in mixed isomer devices correlates with the reduction in overall trap densities as a result of film reorganization to attain thermodynamic equilibrium. We were able to detect with high sensitivity all the intermediate processes occurring at different timescales during the generation/annihilation of electronic states in the band gap of the OSC, including the effects of film transformation from kinetically trapped to thermodynamically stable phases. These results provide access to a remarkable level of detail about the trap dynamics in high mobility OSCs and will not only be instrumental in identifying the processes that hinder the optimal functioning of electronic devices, but can also guide the development of sensing applications that function based on charge carrier trapping. The measurements also offer insights into how the mechanism of charge transport in OSCs is altered by the presence of traps, for example, elucidating the time evolution of band tail states caused by dynamic disorder, the major performance-limiting factor in electronic devices. This type of fundamental advance is instrumental in exploring new pathways for fabricating stable and high-performance devices for the next-generation of large-area and low-cost electronics.

## Conflicts of interest

All authors declare no conflicts of interest.

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## References

- 1 S. Fratini, M. Nikolka, A. Salleo, G. Schweicher and H. Sirringhaus, *Nat. Mater.*, 2020, **19**, 491–502.
- 2 H. Bronstein, C. B. Nielsen, B. C. Schroeder and I. McCulloch, *Nat. Rev. Chem.*, 2020, **4**, 66–77.
- 3 Y. Zhou, C. Fuentes-Hernandez, J. Shim, J. Meyer, A. J. Giordano, H. Li, P. Winget, T. Papadopoulos, H. Cheun, J. Kim, M. Fenoll, A. Dindar, W. Haske, E. Najafabadi, T. M. Khan, H. Sojoudi, S. Barlow, S. Graham, J.-L. Bredas, S. R. Marder, A. Kahn and B. Kippelen, *Science*, 2012, **336**, 327–332.
- 4 B. Russ, A. Glauddell, J. J. Urban, M. L. Chabinye and R. A. Segalman, *Nat. Rev. Mater.*, 2016, **1**, 16050.
- 5 J. Rivnay, S. Inal, A. Salleo, R. M. Owens, M. Berggren and G. G. Malliaras, *Nat. Rev. Mater.*, 2018, **3**, 17086.
- 6 H. F. Haneef, A. M. Zeidell and O. D. Jurchescu, *J. Mater. Chem. C*, 2020, **8**, 759–787.
- 7 M. Nikolka, I. Nasrallah, B. Rose, M. K. Ravva, K. Broch, A. Sadhanala, D. Harkin, J. Charmet, M. Hurhangee, A. Brown, S. Illig, P. Too, J. Jongman, I. McCulloch, J. Bredas and H. Sirringhaus, *Nat. Mater.*, 2017, **16**, 356–362.
- 8 A. Salleo, *Organic Electronics*, Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim, Germany, 2013, pp. 341–380.
- 9 M. F. Calhoun, C. Hsieh and V. Podzorov, *Phys. Rev. Lett.*, 2007, **98**, 096402.
- 10 A. A. Virkar, S. Mannsfeld, Z. Bao and N. Stingelin, *Adv. Mater.*, 2010, **22**, 3857–3875.
- 11 M. J. Ford, J. G. Labram, M. Wang, H. Wang, T.-Q. Nguyen and G. C. Bazan, *Adv. Electron. Mater.*, 2017, **3**, 1600537.
- 12 A. Campos, S. Riera-Galindo, J. Puigdollers and M. Mas-Torrent, *ACS Appl. Mater. Interfaces*, 2018, **10**, 15952–15961.
- 13 P. Westacott, N. D. Treat, J. Martin, J. H. Bannock, J. C. de Mello, M. Chabinye, A. B. Sieval, J. J. Michels and N. Stingelin, *J. Mater. Chem. A*, 2017, **5**, 2689–2700.
- 14 J. Sayago, F. Soavi, Y. Sivalingam, F. Cicoira and C. Santato, *J. Mater. Chem. C*, 2014, **2**, 5690–5694.
- 15 G. Zuo, M. Linares, T. Upreti and M. Kemerink, *Nat. Mater.*, 2019, **18**, 588–593.
- 16 G. Zuo, Z. Li, O. Andersson, H. Abdalla, E. Wang and M. Kemerink, *J. Phys. Chem. C*, 2017, **121**, 7767–7775.
- 17 H. Hirwa and V. Wagner, *Org. Electron.*, 2015, **25**, 112–120.
- 18 M. Nikolka, K. Broch, J. Armitage, D. Hanifi, P. J. Nowack, D. Venkateshvaran, A. Sadhanala, J. Saska, M. Mascal, S.-H. Jung, J. Lee, I. McCulloch, A. Salleo and H. Sirringhaus, *Nat. Commun.*, 2019, **10**, 2122.
- 19 M. Nikolka, G. Schweicher, J. Armitage, I. Nasrallah, C. Jellett, Z. Guo, M. Hurhangee, A. Sadhanala, I. McCulloch, C. B. Nielsen and H. Sirringhaus, *Adv. Mater.*, 2018, **30**, 1801874.
- 20 T. He, Y. Wu, G. D'Avino, E. Schmidt, M. Stolte, J. Cornil, D. Beljonne, P. P. Ruden, F. Würthner and C. D. Frisbie, *Nat. Commun.*, 2018, **9**, 2141.
- 21 T. Hallam, M. Lee, N. Zhao, I. Nandhakumar, M. Kemerink, M. Heeney, I. McCulloch and H. Sirringhaus, *Phys. Rev. Lett.*, 2009, **103**, 256803.
- 22 S. Hunter, A. D. Mottram and T. D. Anthopoulos, *J. Appl. Phys.*, 2016, **120**, 025502.
- 23 K. Takagi, T. Nagase, T. Kobayashi and H. Naito, *Appl. Phys. Lett.*, 2016, **108**, 053305.
- 24 W. L. Kalb and B. Batlogg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 035327.
- 25 R. Häusermann, K. Willa, B. Blülle, T. Morf, A. Facchetti, Z. Chen, J. Lee and B. Batlogg, *Org. Electron.*, 2016, **28**, 306–313.
- 26 M. Geiger, L. Schwarz, U. Zschieschang, D. Manske, J. Pflaum, J. Weis, H. Klauk and R. T. Weitz, *Phys. Rev. Appl.*, 2018, **10**, 044023.
- 27 S. Watanabe, H. Sugawara, R. Häusermann, B. Blülle, A. Yamamura, T. Okamoto and J. Takeya, *Commun. Phys.*, 2018, **1**, 37.
- 28 N. B. Kotadiya, A. Mondal, P. W. M. Blom, D. Andrienko and G. A. H. Wetzelaer, *Nat. Mater.*, 2019, **18**, 1182–1186.
- 29 P. J. Diemer, Z. A. Lampton, Y. Mei, J. W. Ward, K. P. Goetz, W. Li, M. M. Payne, M. Guthold, J. E. Anthony and O. D. Jurchescu, *Appl. Phys. Lett.*, 2015, **107**, 103303.
- 30 C. Krellner, S. Haas, C. Goldmann, K. P. Pernstich, D. J. Gundlach and B. Batlogg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **75**, 245115.
- 31 T. Zimmerling, K. Mattenberger, M. Döbeli, M. J. Simon and B. Batlogg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 134101.
- 32 Z. A. Lampton, K. J. Barth, H. Lee, E. Gann, S. Engmann, H. Chen, M. Guthold, I. McCulloch, J. E. Anthony, L. J. Richter, D. M. DeLongchamp and O. D. Jurchescu, *Nat. Commun.*, 2018, **9**, 5130.
- 33 J. Smith, R. Hamilton, Y. Qi, A. Kahn, D. D. C. Bradley, M. Heeney, I. McCulloch and T. D. Anthopoulos, *Adv. Funct. Mater.*, 2010, **20**, 2330–2337.
- 34 P. J. Diemer, J. Hayes, E. Welchman, R. Hallani, S. J. Pookpanratana, C. A. Hacker, C. A. Richter, J. E. Anthony, T. Thonhauser and O. D. Jurchescu, *Adv. Electron. Mater.*, 2017, **3**, 1600294.
- 35 R. K. Hallani, K. J. Thorley, Y. Mei, S. R. Parkin, O. D. Jurchescu and J. E. Anthony, *Adv. Funct. Mater.*, 2016, **26**, 2341–2348.
- 36 D. Lehnher, A. R. Waterloo, K. P. Goetz, M. M. Payne, F. Hampel, J. E. Anthony, O. D. Jurchescu and R. R. Tykwinski, *Org. Lett.*, 2012, **14**, 3660–3663.
- 37 C.-W. Huang, X. You, P. J. Diemer, A. J. Petty, J. E. Anthony, O. D. Jurchescu and J. M. Atkin, *Commun. Chem.*, 2019, **2**, 22.
- 38 M. Grünewald, P. Thomas and D. Würtz, *Phys. Status Solidi*, 1980, **100**, K139–K143.



- 39 W. L. Kalb, F. Meier, K. Mattenberger and B. Batlogg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 184112.
- 40 W. L. Kalb, S. Haas, C. Krellner, T. Mathis and B. Batlogg, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 155315.
- 41 M. Waldrup, O. D. Jurchescu, D. J. Gundlach and E. G. Bittle, *Adv. Funct. Mater.*, 2020, **30**, 1904576.
- 42 W. H. Lee, D. H. Kim, J. H. Cho, Y. Jang, J. A. Lim, D. Kwak and K. Cho, *Appl. Phys. Lett.*, 2007, **91**, 092105.
- 43 H. Ullah Khan, R. Li, Y. Ren, L. Chen, M. M. Payne, U. S. Bhansali, D.-M. Smilgies, J. E. Anthony and A. Amassian, *ACS Appl. Mater. Interfaces*, 2013, **5**, 2325–2330.
- 44 K. C. Dickey, J. E. Anthony and Y.-L. Loo, *Adv. Mater.*, 2006, **18**, 1721–1726.
- 45 Z. A. Lamport, H. F. Haneef, S. Anand, M. Waldrup and O. D. Jurchescu, *J. Appl. Phys.*, 2018, **124**, 071101.
- 46 S. Park, S. H. Kim, H. H. Choi, B. Kang and K. Cho, *Adv. Funct. Mater.*, 2020, **30**, 1904590.
- 47 J. Sworakowski, *Brazilian J. Phys.*, 1999, **29**, 318–331.