RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This Accepted Manuscript will be replaced by the edited, formatted and paginated article as soon as this is available.

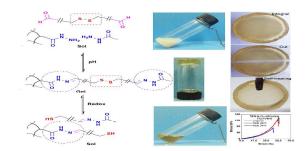
You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



1

Table of contents entry



2

- 3 Redox- and pH-responsive polymer gel with self-healing property was prepared by crosslinking of
- 4 benzhydrazide-containing polytriazole with a disulfide-containing dialdehyde.

5

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

Redox- and pH-responsive polymer gels with reversible sol-gel transitions and self-healing properties

Ping Zhang^a, Fengyang Deng^a, Ya Peng^a, Hongbiao Chen*^a, Yong Gao^a, Huaming Li*^{a,b}

5 Received (in XXX, XXX) Xth XXXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX DOI: 10.1039/c0xx00000x

Covalent dynamic gels based on reversible acylhydrazone and disulfide bonds were prepared by crosslinking of benzhydrazidecontaining poly(triazole) (PTB) with a novel disulfide-containing dialdehyde (C2) in DMF at ambient temperature. The PTB was synthesized by the metal- and solvent-free click polymerization of diazide and dialkyne monomers followed by treatment with hydrazine 10 hydrate. The as-fabricated polymer gels exhibited both redox and pH stimuli-responsive behaviours. Analysis of the compositionproperty relationships of these polymer gels, specifically considering the effects of catalysts, molar ratio of benzhydrazide to aldehyde groups (-NH₂/-CHO), and gelator concentrations on rheological properties, were performed. Additionally, the gel revealed interesting self-healing property through acylhydrazone exchange and/or disulfide exchange reactions. Employing this dynamic character, it is possible to regenerate the used gel, and thus has the potential to perform in a range of dynamic or bioresponsive applications.

15 1. Introduction

Stimuli-responsive polymer gels have recently been the focus of fundamental and applied researches because they have potential applications in controlled drug delivery devices, sensors, sensors, applications actuators,³ and biomaterials.⁴ Essentially, stimuli-responsive gels

- 20 can dynamically and reversibly undergo a sol-gel transition in response to changes in their environments, therefore altering their structures and properties. Up to now, two effective strategies, including supramolecular chemistry (noncovalent interactions)⁵⁻⁷ and dynamic covalent chemistry (dynamic covalent bonds)8-10
- 25 have been developed to fabricate such polymeric networks. Although supramolecular chemistry is a simple way to fabricate stimuli-responsive polymer gels, it is less robust and the resultant gels are usually labile and do not have sufficient mechanical strength, thus may not be suitable for all applications. In contrast,
- 30 dynamic covalent chemistry relates to chemical reactions carried out reversibly under conditions of equilibrium control. 11,12 These bonds become reversible and reach an equilibrium state once they are exposed to appropriate external stimuli such as pH,13 light,14 redox, 15 temperature, 16 biomolecules, 17 and the addition of
- 35 catalysts. 18 Polymer gels crosslinked with dynamic covalent bonds would provide an energetically favorable, specific, and controlled mechanism for engineering functional dynamic networks. The most significant advantage of such polymer gels is that they are not only flexible in altering their compositions, as in
- 40 the case of supramolecular polymers with noncovalent interactions, but also stable enough to maintain their structure in the absence of external stimuli, as in the case of conventional covalent polymers.¹⁹ Therefore, dynamic covalent chemistry offers a robust platform to construct stimuli-responsive polymer
- 45 gels, which integrates the stability of the chemical gels and stimuli-responsibility of the physical gels into a single system.²⁰

So far various dynamic covalent chemistries, such as exchange over C=N double bond,21 disulfide exchange,22 radical exchange

reaction,²³ trans-esterification,²⁴ Diels-Alder cycloaddition,²⁵ 50 boronate ester bond, ^{26,27} and olefin metathesis, ²⁸ have been used to prepare stimuli-responsive polymer gels. Among these, C=N bond in acylhydrazone groups formed by the condensation of hydrazides with carbonyl group exhibit reversibility under mild conditions.²⁹ Generally, the pH value of the solution significantly 55 affects the reactivity of the C=N bond, which become relatively stable in alkaline solution, while higher acidity will destroy this bond. With such characteristics, the C=N bond can be used as a good stimulus-responsive linker in polymer gels. For example, Deng and his coworkers²⁰ employed a triple aldehyde derivate to 50 crosslink bisacylhydrazine modified poly(ethylene oxide) in organic solvents, yielding dynamic polymer gels with reversible sol-gel phase transitions and self-healing properties based on the reversible breaking and regenerating of acylhydrazone bonds. However, simply mixing the polymer and crosslinker was useless, 55 only after adding glacial acetic acid, adjusting the apparent pH to 6-7, did gelation finally occur. In a similar way, chitosan based hydrogel was successfully constructed by crosslinking of chitosan with benzaldehyde modified poly(ethylene oxide).³⁰ The as-

Comparable with the acylhydrazone bond, disulfide exchange³¹ is perhaps another promising reaction for dynamic covalent chemistry, which is multi-responsive to pH, 13 light, 32 and redox 33 75 conditions. Mechanistic studies indicate that thiols can readily oxidize to disulfide bond upon exposure to air. On the other hand, disulfide bond can be cleaved under mild conditions in the presence of reductants such as dithiothreitol (DTT), tris(2carboxyethyl)phosphine (TCEP), and glutathione (GSH).³⁴ As a 30 matter of fact, disulfide bond has been successfully used to prepare responsive polymer gels from small molecular gelators³⁴ or macromolecular gelators.35 Very recently, multi-responsive polymer gels were fabricated using disulfide bond in conjunction

formed gel shows its response to changes of the pH value, and

gelation still exists after six cycles. The self-healing performance

of the hydrogel was also outstanding.

with acylhydrazone bond, since they have been demonstrated to be compatible with each other and independently operational upon pH changes in a single system. However, so far as we know there is only one report describing the construction of dual-5 responsive polymer gels based on disulfide and acylhydrazone bonds, in which benzlaldehyde-terminated 3-armed poly(ethylene oxide) was crosslinked with dithiodipropionic acid dihydrazide. As can be seen, only a small percentage of known polymers so far have been used to construct such multi-stimuli responsive 10 polymer gels, thereby placing a restriction on the range of chemophysical properties and potential applications.

In this paper, we report a novel dual-responsive polymer gels based on acylhydrazone and disulfide bonds, in which the benzhydrazide-containing poly(triazole) (PTB) was crosslinked 15 with a novel disulfide-containing dialdehyde in DMF at ambient temperature. The PTB was synthesized by the metal- and solventfree click polymerization of diazide and dialkyne monomers followed by treatment with hydrazine hydrate.³⁷ The as-fabricated gels exhibited both redox and pH stimuli-responsive behaviors 20 due to the incorporation of the two dynamic bonds into the same polymer networks (Scheme 1). Additionally, reversible crosslinks allow these networks to restructure dynamically and self-healing after mechanical disruption. Being responsive to multiple stimuli, these polymer gels are expected to be highly advantageous over 25 traditional polymer gels and possess many unique properties that play a significant role in organ repairs, drug-delivery systems, and molecular devices.38

2. Experimental

2.1 Materials

30 Well-defined poly(triazole)s (PTAs) were prepared by the metaland solvent-free, thermal click polymerization of methyl 3,5dipropargyloxylbenzoate and methyl 3,5-bis(4-azidobutoxy)
benzoate monomers followed by multistep precipitation in
different precipitating agents as reported previously by our
35 group³⁷ (see Supporting Information, SI). The obtained PTAs
were further subjected to postfunctionalization and were treated
with hydrazine hydrate, yielding benzhydrazide-containing PTAs
(PTB) (see SI). The crosslinker, 4,4-[4,4'-dithiobis(1-butoxyl)]
diphenylformaldehyde (C2), was prepared according to the
40 literature procedure³⁹ as shown in Scheme S1. All other reagents
were purchased from commercial supplier, which were used as
received without further purification.

2.2 Typical procedure for the preparation of gels

By condensation of benzaldehyde groups at the two ends of C2 45 with benzhydrazide moieties in PTB, a network with acylhydrazone bond as crosslinks is generated. A desired amount of PTB and C2 was dissolved separately in DMF, the appropriate amount of catalyst was added. Gels were formed simply by pipetting the prepared solutions directly into a vial at room 50 temperature. Variable polymer gels prepared by adjusting catalysts (types and contents), molar ratio of benzhydrazide to benzaldehyde moieties (–NH₂/–CHO), and gelator concentrations, were summarized in Table S1.

2.3 Rheological measurements

Dynamic moduli measurements were carried out with a stress-controlled Advance Rheological Expansion System (ARES) equipped with a parallel plate fixture of a diameter of 25 mm 50 diameter at 25 °C. The preliminary dynamic strain sweep was carried out first at angular frequency of 1 rad/s to determine the linear viscoelastic region and then the frequency sweep was performed over the frequency range of 0.01 to 100 rad/s at strain of 1%. And 48 h were taken from mixing the gelator solutions to 55 testing start. Both elastic modulus (G') and loss modulus (G") of these polymer gels prepared with different catalysts and contents, —NH₂/–CHO, and gelator concentrations were monitored as a function of frequency to characterize the viscoelastic properties.

2.4 Compression tests

70 The polymer gel was synthesized in a cylinder that was 10 mm in diameter and 10 mm in height. The compression test was carried out using a mechanical testing apparatus (Instron 5943, Instron Corp.) under uniaxial compression using parallel plate geometry. Test parameters were as follows: load cell 1 kN, crosshead 75 velocity was 1 mm/min. Before loading on to the Instron, the samples were equilibrated at ambient temperature for 48 h. Mechanical compression data were averaged over three samples.

2.5 Self-healing experiments

Polymer gel plates were prepared in a mold with a bottom of 25 30 mm in diameter. After aged in the mold at room temperature for about 48 h, each of the gel plates was cut by a blade into two halves. Then the two pieces were spliced in the original mold, and the mold was placed in a container at ambient temperature without any other intervention. After 48 h, the gel plate(s) was 35 taken out of the mold and was drawn perpendicularly to the cut lines by tweezers, with picture taken.

3. Results and discussion

3.1 Synthesis of gelators

In order to synthesize polymers containing benzhydrazide groups, 90 benzoate-containing PTAs were produced initially. In this paper, we recount significant milestone in achieving metal-free thermal click polymerization to synthesize polymers. Firstly, a series of well-defined PTAs were prepared by the metal- and solvent-free, thermal click polymerization of methyl 3,5-dipropargyloxyl 35 benzoate and methyl 3,5-bis(4-azidobutoxy)benzoate monomers followed by multistep precipitation in different precipitating agents (Fig. S1) as reported previously by our group.³⁷ PTA with molecular weight of 32430 g/mol (PDI = 1.22, Fig. S2) was then subjected to postfunctionalization and was treated with hydrazine)0 hydrate under reflux, in which the benzoate groups were easily converted to benzhydrazide moieties,40 yielding benzhydrazidecontaining poly(triazole) (PTB). The functionalization of polymer was confirmed using ¹H NMR spectroscopy. As depicted in Fig. S3, a new peak at about δ 9.72 ppm, due to amide protons in the)5 benzhydrazide moiety, 41 was observed, while the peak at δ 3.79 ppm due to methyl protons in the benzoate moiety completely disappeared, confirming that the functionalization reaction was carried out essentially to quantitative conversion.

Next, the crosslinker C2 was synthesized according to the synthetic routes as shown in Scheme S1. Firstly, the 4hydroxybenzaldehyde reacted with excess 1,4-dibromobutane to introduce bromide groups. Secondly, the halogen group was 5 transformed into thiol group by treatment with freshly prepared sodium hydrosulfide. Crosslinker C2 was finally synthesized by the oxidation of thiol group in air. The crosslinker was well characterized by IR, MS, and NMR spectroscopy (Fig. S4-S9).

3.2 Preparation of stimuli-responsive polymer gels

10 As mentioned previously, the primary motivation of this study is to prepare redox- and pH-responsive PTA gels with reversible sol-gel transitions together with self-healing properties. Previous studies have demonstrated that the triazole repeat units of PTAs are structurally similar to the amide bond of native peptide in 15 terms of distance and planarity, 42 and their biocompatibility may

Scheme 1. Strategy for constructing a dynamic polymer gel with dual responsive sol-gel transitions based on acylhydrazone and disulfide chemistry.

allow PTA based gels to find a wide variety of applications in regulate cell culture, 43 bioconjugates, 4 biocontainers, 44 and 30 biocarriers. 45 Considering that acvlhydrazone bond formation can be accelerated under mild acidic conditions or by catalytic aniline in neutral conditions, gelation conditions of PTB ($M_n = 30560$ g/mol, PDI = 1.26, Fig. S2, sic passim) plus C2 in DMF were initially explored with different catalysts, i.e., glacial acetic acid 35 (HOAc) and aniline (PhNH₂). Desired amounts of C2/catalyst mixture as well as PTB were separately dissolved in DMF with different catalysts or catalyst contents. As can be seen in Table S1, gel comprising of 15 wt% gelators was formed immediately in the presence of 20 wt% HOAc. However, the gelation time, the 40 interval of time required for the mixed solutions to become a gel,

was proved to be dependent on the catalyst contents and related with the used catalysts (Table S1). Decrease the contents of HOAc resulted in the prolongation of gelation time. On the other hand, the gelation time was prolonged to 9 h when the catalyst

45 was changed from HOAc (10 wt%) to PhNH₂ (10 wt%) and further prolonged to 17 h without catalyst.

In addition, the gelation time was also proved to be correlated with the -NH₂/-CHO molar ratios and gelator concentrations (Table S1). When the gelator concentration and HOAc content 50 were fixed at 15 wt% and 10 wt%, respectively, gels formed quickly within 15 min at room temperature with the -NH₂/-CHO ratios ranged from 12.5/1 to 5/1. Similarly, gel formed within 20 min when the gelator concentration was varied from 8 to 20 wt%. However, no gel formed in 10 days when gelator concentration 55 was lower than 5 wt%. These results indicate that the reactivity of benzhydrazide and benzlaldehyde can be tuned by catalyst types as well as catalyst and gelator concentrations.

FTIR analysis provided evidence for the formation of covalent bond between benzhydrazide and benzlaldehyde groups in the gel. 60 Fig. 1 shows the FTIR spectra of C2, PTB, and the dried gel. The

characteristic carbonyl infrared absorbances for benzlaldehyde and benzhydrazide groups are observed at 1693 and 1662 cm⁻¹, respectively. For polymer gel, the carbonyl of acylhydrazone group slightly shifts to 1667 cm⁻¹, indicating that benzhydrazide 55 group has reacted with benzlaldehyde group.²⁹ This result further proves that the gel is a chemical gel. Moreover, the absorbance of benzhydrazide groups (i.e., 1662 cm⁻¹) remains relatively strong in the gel by comparing the FTIR spectra of PTB (spectrum b) and the polymer gel (spectrum c), indicating that certain amount 70 of free benzhydrazide groups are preserved in the gel, which favor further reactions in self-healing.

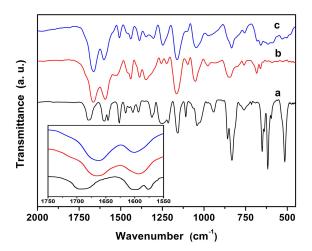


Fig. 1 FTIR spectra of (a) C2, (b) PTB, and (c) polymer gel.

3.3 Rheological characterization and stress-strain behaviors

75 In an attempt to gain a clearer understanding of the viscoelastic properties of the PTA based gel samples, dynamic rheological measurements were therefore performed. The effect of catalysts was initially studied. Fig. 2 shows the dynamic frequency sweeps of polymer gels prepared with either 10 wt% of HOAc, 10 wt% of PhNH₂, or without catalyst, while the gelator concentration 5 and the –NH₂/–CHO molar ratio are kept constant, *i.e.*, 15 wt% and 10/1, respectively. For the three gel samples, G' exceeded G' over the entire measured frequency and G' was independent of angular frequency (Fig. 2a), implying their predominantly solid-like behavior. These are the characteristics for a strong 10 polymer gel with covalent crosslinked networks. However, gel prepared with 10 wt% HOAc possessed the largest G', while gel without catalyst possessed the minimum G', indicating that HOAc can accelerate the reaction between benzlaldehyde and benzhydrazide, therefore increasing the crosslinking density.

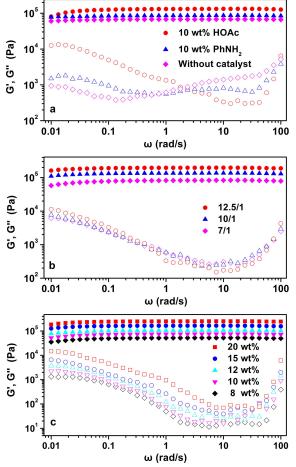


Fig. 2 G' (solid symbol) and G" (open symbol) against frequency (ω) for polymer gels with (a) different catalysts (-NH₂/-CHO = 10/1, gelator concn. = 15 wt%); (b) different -NH₂/-CHO molar ratios (gelator concn. = 15 wt%, HOAc concn. = 10 wt%); and (c) 20 different gelator concn. (-NH₂/-CHO = 10/1, HOAc concn. = 10 wt%).

The viscoelastic properties of the PTA polymer gels can also be affected by the molar ratio of –NH₂/–CHO (Fig. 2b). Similarly, 25 G' is higher than G" over the entire frequency range for all polymer gels, further indicating that the elastic response of these gels is stronger than the viscous one. Although a stiffer and stronger gel network can be obtained at the –NH₂/–CHO molar

ratio of 12.5/1, higher G' is observed with increasing the 30 aldehyde contents due to the increase of crosslinking density in polymer networks. The viscoelastic properties of the polymer gels were further studied by varying gelator concentrations (Fig. 2c). As expected, G' increased with increasing the gelator concentration due to an increase in crosslink density. Again, G' 35 was found to exceed G" over the entire measured frequency and G' was independent of angular frequency. Similar results were found for polymer gels prepared with different HOAc contents as well as with different PTB molecular weights (Fig. S10-S11).

The effects of gelator concentration on the compressive stress-40 strain characteristics of the polymer gels are shown in Fig. 3. The corrected true stress and strain were calculated as FH/A₀H₀ and $-\ln(H/H_0)$, where H was the varying sample height, H_0 its original height, A_{θ} its original cross-section area, and F the applied load. 48,49 Clearly, the compressive stress of the polymer 45 gels increases with the increase of gelator concentrations at the same strain level. This effect is more distinct with the rise of strain ratio. This result is not surprising in view of the increased modulus of polymer gels with increasing gelator concentrations as discussed previously. The compression moduli (E) estimated 50 from the linear region (6–18%) of the stress–strain curves are tabulated in Table S3. As can be seen, the E values increase with increasing of gelator concentrations, which mirrors the trend observed for the dynamic rheology data in Fig. 2c (the G' values estimated from Fig. 2c are also shown in Table S3). However, 55 exact agreement between the modulus values obtained from dynamic rheology and static compression testing experiments cannot be expected because of the effect of the Poisson ratio and differences in measurement conditions (especially compression

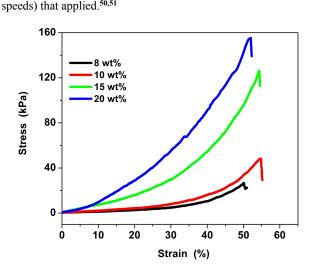


Fig. 3 Stress-strain curves for polymer gels with different gelator concentrations ($-NH_2/-CHO = 10/1$, HOAc concn. = 10 wt%).

3.4 Sol-gel transitions and self-healing properties

Due to the incorporation of acylhydrazone and disulfide dynamic 55 bonds into the same polymer networks, the polymer gels should display reversible sol-gel transitions in response to both pH and redox conditions (Scheme 1). In order to prove the reversible solgel phase transition ability of the gel in response to pH change, polymer gel prepared in 0.5 mL of DMF (–NH₂/–CHO = 10/1,

gelator concn. = 10 wt%, without catalyst) was aged for 48 h at ambient temperature. After 60 µL of CF₃COOH (TFA) was added on the gel surface, the gel was degelated into a sol in 12 h. And 20 μL of Et₃N (TEA) was needed to transform the sol into gel in 5 12 days (Fig. 4a). The reversible sol-gel transitions was triggered by reversible breaking and regenerating of acylhydrazone bonds upon pH changes.⁵² In the current case, however, it took so long to return to the gel state, indicating that the regeneration of acylhydrazone bonds was very slow due to the poor solubility of 10 polymer in TFA as well as the undissolved salts formed from TFA and TEA. For the same reason, the sol as well as the reformed gel became turbid and the transition process could be run only once.

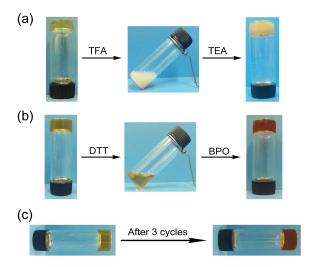


Fig. 4 Sol-gel phase transition of the polymer gels in response to 15 pH (a) and redox (b) triggers, and (c) appearance of the original gel (left) and the gel after three redox cycles (right).

To further prove the redox responsive sol-gel phase transitions of the polymer gels, three gel samples were separately prepared in 0.5 mL of DMF containing either 10 wt% of HOAc, 10 wt% of 20 PhNH₂, or without catalyst (Table S1, run 1, 3, and 6) and aged for 48 h. As expected, three transition cycles were repeated with using dithiothreitol (DTT) and benzoyl peroxide (BPO) as the reductant and oxidant, respectively (Fig. 4b, Table S2). For each transition cycle, DTT (30 mg) and BPO (50 mg) were added in 25 sequence. In addition, Et₃N was firstly added to the polymer gels to guarantee alkaline condition for the reduction of disulfide bond, because the disulfide-thiol redox reaction is switched-off under slightly acidic conditions.⁵³ For the polymer gel prepared with HOAc catalyst, about 50 μL of Et₃N was added and part of it was 30 used to neutralize the HOAc in polymer gel system. During the repeated transition cycles, the polymer gels turn into deep red in color due to the fact that the PTB solution in DMF gradually became brown in air (Fig. 4c, Fig. S12).

In general, dynamic covalent systems have enhanced stability 35 while maintaining the potential for dynamics and self-healing as compared to supramolecular assembly that rely on intermolecular forces. The adaptive self-healing⁵⁴⁻⁵⁶ properties of these polymer gels, based on the dynamic chemistry of both acylhydrazone and disulfide bonds, were further studied and the results are presented 40 in Fig. 5. Again, three gel samples were separately prepared in 1.5 mL of DMF containing either 10 wt% of HOAc, 10 wt% of

PhNH₂, or without catalyst (Table S1, run 1, 3, and 6) and aged for 48 h. As shown in Fig. 5, when two pieces of cracked gel plates were put together along the cut line and kept contact f at 40 45 °C or 48 h in the absence of external intervention, they can merge autonomously into a whole piece of gel. The joint between the two parts in the merged gel was strong enough to be lifted by the tweezers (Fig. 5, 1c-3c). The mechanism of self-healing process should be related with the disulfide-disulfide exchange as well as 50 acylhydrazone–acylhydrazone exchange reactions.^{57,58} When the cut surfaces contact each other and within this state, the polymer segments can diffuse and cross the interface of two plates and reactions such as disulfide exchange and acylhydrazone exchange can occur, therefore regenerating acylhydrazone and/or disulfide 55 bonds between the cut edges and eventually healing the gel. Obviously, the chain segments in the network need mobility to allow for the interchange reactions to take place.

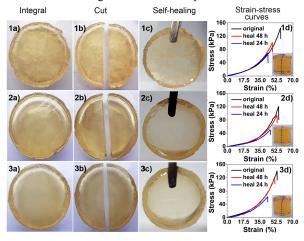


Fig. 5. (a-c) Self-healing properties of polymer gels. (d) Stress-50 strain curves of the three self-healable polymer gels. 1, 2, and 3 represent polymer gels prepared without catalyst, 10% HOAc, and 10% PhNH₂, respectively, -NH₂/-CHO = 10/1, gelator concn. = 15 wt%. Inset of d shows samples for stress-train measurements, black dotted line represents the location of the 55 incision.

In order to prove this self-healing process was repeated and effective, the merged gel was subjected to compressive stressstrain measurements and healing efficiency is defined as the ratio of the stress at break of healed and virgin gels.⁵⁹ For the polymer 70 gel prepared without catalyst, the healing efficiency is around 70.6% after 48 h of self-healing (Fig. 5, 1d), due to the fact that both acylhydrazone and disulfide bonds reconstruct slowly in organic solvent under neutral conditions. In addition, the healing efficiency for 48 h is higher than that for 24 h, suggesting that 75 longer self-healing time leads to better healing. Under acidic condition, the acylhydrazone exchange reaction is responsible for self-healing process because disulfide bonds are kinetically locked. 55 In the case of HOAc catalyzed polymer gel, the healing efficiency is around 80.6% after 48 h of self-healing (Fig. 5, 3d). 30 For the PhNH₂ catalyzed gel, a maximum healing efficiency of 85% is observed after 48 h of healing since both acylhydrazone exchange and disulfide exchange reactions are responsible for this self-healing process. In the present study, repetitive healing experiments were also conducted. After being cut again along the 35 old cut lines, the healing efficiencies after the second and third

healing process are fully restored.

4. Conclusions

In summary, we have prepared covalent dynamic poly(triazole) gels containing both acylhydrazone and disulfide bonds by 5 crosslinking of benzohydrazide-containing poly(triazole) with disulfide-containing dialdehyde in DMF at ambient temperature. The gelation time is proved to be dependent on the catalyst types and contents and related with the molar ratio of benzohydrazide to aldehyde moieties as well as the gelator concentrations. In

- 10 addition, the orthogonal dynamic chemistry of the two bonds makes the as-synthesized polymer gels self-healable under a wide range of conditions and the self-healing process is repetitive and effective without an external stimulus. Moreover, the polymer gels also display reversible sol-gel transitions in response to both
- 15 pH and redox conditions. Being responsive to multiple stimuli, these polymer gels are expected to be highly advantageous over traditional polymer gels and possess many unique properties that play a significant role in organ repairs, drug-delivery systems, and molecular devices.

20

Acknowledgements

Financial support from the Program for NSFC (51273170), and the Construct Program of the Key Discipline in Hunan Province 30 is greatly acknowledged.

25 Notes and references

- ^a College of Chemistry, Xiangtan University, Xiangtan, Hunan Province,
- ^b Key Laboratory of Polymeric Materials & Application Technology of Hunan Province, and Key Laboratory of Advanced Functional Polymeric
- 30 Materials of College of Hunan Province, Xiangtan 411105, Hunan Province, PR China.

Fax: +86 731 58293264; Tel: +86 731 58298572;

Email:lihuaming@xtu.edu.cn

- † Supporting Information (SI) available: Synthesis and characterization 35 of PTAs, PTB, and crosslinker C2; The effect of different catalysts, catalyst contents, -NH₂/-CHO molar ratio, and gelator concentrations on the gelation time in DMF; The redox responsive sol-gel transitions of polymer gels; G' and G" against frequency for polymer gels with different HOAc contents as well as gels prepared from PTB with different
- 40 molecular weights. See DOI: 10.1039/b000000x/
 - ‡ Footnotes should appear here.
- 2. O. Bunkoed, F. Davis, P. Kanatharana, P. Thavarungkul and S. P. Higson, Anal. Chim. Acta, 2010, 659, 251-257.
 - 3. N. Terasawa, N. Ono, Y. Hayakawa, K. Mukai, T. Koga, N. Higashi and K. Asaka, Sens. Actuators, B, 2011, 160, 161-167.
 - 4. E. Ye, P. L. Chee, A. Prasad, X. T. Fang, C. Owh, V. J. J. Yeo and X. J. Loh, Mater. Today, 2014, 17, 194-202.
- 50 5. J.-M. Lehn, Chem. Soc. Rev., 2007, 36, 151-160.
 - J.-M. Lehn, Prog. Polym. Sci., 2005, 30, 814-831.
 - 7. U. P. Singh, S. Kashyap, H. J. Singh, B. K. Mishra, P. Roy and A. 25 Chakraborty, J. Mol. Struct., 2012, 1014, 47-56.
 - S. Otto, R. L. Furlan and J. K. Sanders, Drug Discov. Today, 2002, 7, 117-125.
 - S. J. Rowan, S. J. Cantrill, G. R. L. Cousins, J. K. M. Sanders and J. F. Stoddart, Angew. Chem. Int. Ed., 2002, 41, 898-952.

- 10. T. Yang, R. Ji, X. X. Deng, F. S. Du and Z. C. Li, Soft matter, 2014, 10, 2671-2678.
- 50 11. H. Otsuka, K. Aotani, Y. Higaki and A. Takahara, J. Am. Chem. Soc., 2003, 125, 4064-4065.
 - 12. W. Zhang and J. S. Moore, J. Am. Chem. Soc., 2004, 126, 12796-12796
 - 13. I. Asmarandei, G. Fundueanu, M. Cristea, V. Harabagiu and M. Constantin, J. Polym. Res., 2013, 20, 1-13.
 - 14. Y. Guan, H. B. Zhao, L. X. Yu, S. C. Chen and Y. Z. Wang, RSC Adv., 2014, 4, 4955-4959.
 - 15. Z. Y. Qiao, R. Zhang, F. S. Du, D. H. Liang and Z. C. Li, J. Control. Release, 2011, 152, 57-66.
- 70 16. P. P. Li, L. Huang, Y. J. Lin, L. Shen, Q. Chen and W. Z. Shi, Nanotechnology, 2014, 25, 055603.
 - 17. T. Miyata, T. Uragami and K. Nakamae, Adv. Drug Deliver. Rev., 2002, 54, 79-98.
- 18. Z. S. Ge, D. Xie, D. Y. Chen, X. Z. Jiang, Y. F. Zhang, H. W. Liu and S. Y. Liu, Macromolecules, 2007, 40, 3538-3546.
- 19. R. J. Wojtecki, M. A. Meador and S. J. Rowan, Nat. Mater., 2011, 10, 14-27.
- 20. G. H. Deng, C. M. Tang, F. Y. Li, H. F. Jiang and Y. M. Chen, Macromolecules, 2010, 43, 1191-1194.
- 30 21. J. A. Nam, A. Al-Nahain, S. Hong, K. D. Lee, H. Lee and S. Y. Park, Macromol. Biosci., 2011, 11, 1594-1602.
 - 22. B. Gyarmati, Á. Némethy and A. Szilágyi, RSC Advances, 2014, 4,
- 23. H. Y. Pan, Y. Y. Zhang, H. T. Pu and Z. H. Chang, J. Power Sources, 35 2014, 263, 195-202.
 - 24. M. Babazadeh, J. Appl. Polym. Sci., 2007, 104, 2403-2409.
 - 25. Z. Wei, J. H. Yang, X. J. Du, F. Xu, M. Zrinyi, Y. Osada, F. Li and Y. M. Chen, Macromol. Rapid Commun., 2013, 34, 1464-1470.
- 26. J. Xu, D. G. Yang, W. J. Li, Y. Gao, H. B. Chen and H. M. Li, Polymer, 2011, 52, 4268-4276.
 - 27. W. J. Li, M. N. Liu, H. B. Chen, J. Xu, Y. Gao and H. M. Li, Polym. Adv. Technol., 2014, 25, 233-239.
 - 28. J. S. Kingsbury, S. B. Garber, J. M. Giftos, B. L. Gray, M. M. Okamoto, R. A. Farrer, J. T. Fourkas and A. H. Hoveyda, Angew. Chem. Int. Ed., 2001, 40, 4251-4256.
 - 29. L. J. Zhu, Y. F. Shi, C. L. Tu, R. B. Wang, Y. Pang, F. Qiu, X. Y. Zhu, D. Y. Yan, L. He, C. Y. Jin and B. S. Zhu, Langmuir, 2010, 26, 8875-8881
- 30. E. Lih, J. S. Lee, K. M. Park and K. D. Park, Acta Biomater., 2012, 8, 3261-3269.
 - 31. R. Bird, T. Freemont and B. R. Saunders, Soft Matter, 2012, 8, 1047-
- 32. Y. Koyama, T. Yoshii, Y. Kohsaka and T. Takata, Pure Appl. Chem., 2013, 85, 835-842.
- 33. H. J. Zhang, L. Peng, Y. Xin, Q. Yan and J. Y. Yuan, Macromol. Symp., Wiley Online Library, 2013, 329. 66-69.
 - 34. C. H. Ren, Z. J. Song, W. T. Zheng, X. M. Chen, L. Wang, D. L. Kong and Z. M. Yang, Chem. Commun., 2011, 47, 1619-1621.
- 35. S. C. Tang, M. J. Glassman, S. L. Li, S. Socrate and B. D. Olsen, Macromolecules, 2014, 47, 791-799.
- 36. G. H. Deng, F. Y. Li, H. X. Yu, F. L. Liu, C.Y. Liu, W. X Sun, H. F.
- Jiang and Y. M. Chen, ACS Macro Lett., 2012, 1, 275-279. 37. F. Y. Deng, B. Xu, Y. Gao, Z. Liu, D. G. Yang and H. M. Li, J. Polym.
- Sci., Part A: Polym. Chem., 2012, 50, 3767-3774. 1. N. Bhattarai, J. Gunn and M. Q. Zhang, Adv. Drug Deliver. Rev. 2010, 15 38. A. P. Zhang, Z. Zhang, F. H. Shi, J. X. Ding, C. S. Xiao, X. L. Zhuang, C. L. He, L. Chen and X. S. Chen, Soft Matter, 2013, 9,
 - 2224-2233. 39. Q. Huang and B. Zheng, Org. Chem. Int., 2012, 2012, doi:10.1155/2012/208128.
 - 20 40. J. Amartey, C. Esguerra, B. Al-Otaibi, R. Parhar and I. Al-Jammaz, Appl. Radiat. Isot., 2004, 60, 839-843.
 - 41. A. Alborzi, S. Zahmatkesh and A. Yazdanpanah, Polym. Bull., 2013, **70**, 3359-3372
 - 42. M. Ghaemy, S. Qasemi, K. Ghassemi and M. Bazzar, J. Polym. Res., 2013, 20, 1-15.
 - 43. D. Hazar Apaydın, D. Esra Yıldız, A. Cirpan and L. Toppare, Sol. Energy Mater. Sol. Cells, 2013, 113, 100-105.
 - 44. R. Bagtache, G. Rekhila, K. Abdmeziem and M. Trari, Mater. Sci. Semicond. Process., 2014, 23, 144-150.

- 45. M.-K. Song, X. B. Zhu and M. L. Liu, J. Power Sources, 2013, 241, 219-224.
- 46. J. W. Steed, Chemical Society Reviews, 2010, 39, 3686-3699.
- 47. Y. Lee, H. J. Chung, S. Yeo, C.-H. Ahn, H. Lee, P. B. Messersmith, and T. G. Park, Soft Matter, 2010, 6, 977-983.
- 48. I. Arvanitoyannis, E. Psomiadou and A. Nakayama, Carbohydr. Polym., 1996, 31, 179-192.
- 49. E. Amici, A. Clark, V. Normand and N. Johnson, Biomacromolecules, 2000, 1, 721-729.
- 10 50. T. Lane, J. L. Holloway, A. H. Milani, J. M. Saunders, A. J. Freemont and B. R. Saunders, Soft Matter, 2013, 9, 7934-7941
 - 51. D. J. Adams, L. M. Mullen, M. Berta, L. Chen and W. J. Frith, Soft Matter, 2010, 6, 1971-1980.
 - 52. J. Y. Li, S. X. Yang, L. Wang, X. B. Wang and L. Liu, Macromolecules, 2013, 46, 6832-6842.
 - 53. Z. Rodriguez-Docampo and S. Otto, Chem. Commun., 2008, 5301-
 - 54. U. Gulyuz and O. Okay, Soft Matter, 2013, 9, 10287-10293.
 - 55. R. P. Wool, Soft Matter, 2008, 4, 400-418.
- 20 56. G. Akay, A. Hassan-Raeisi, D. C. Tuncaboylu, N. Orakdogen, S. Abdurrahmanoglu, W. Oppermann and O. Okay, Soft Matter, 2013, 9, 2254-2261.
 - 57. J. A. Yoon, J. Kamada, K. Koynov, J. Mohin, R. Nicolay, Y. Z. Zhang, A. C. Balazs, T. Kowalewski and K. Matyjaszewski, Macromolecules, 2012, 45, 142-149.
 - 58. J. A. Syrett, C. R. Becer, D. M. Haddleton, Polym. Chem, 2010, 1, 978-987.
 - 59. A. M. Peterson, R. E. Jensen and G. R. Palmese, ACS Appl. Mater. Inter., 2010, 2, 1141-1149.