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Infrared photodissociation spectroscopy of $(\text{Al}_2\text{O}_3)_{2-5}\text{FeO}^+$: influence of Fe-substitution on small alumina clusters†

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The infrared photodissociation spectra of He-tagged $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ ($n = 2-5$), are reported in the Al–O and Fe–O stretching and bending spectral region (430–1200 cm^{-1}) and assigned based on calculated harmonic IR spectra from density functional theory (DFT). The substitution of Fe for an Al center occurs preferentially at 3-fold oxygen coordination sites located at the cluster rim and with the Fe atom in the +III oxidation state. The accompanying elongation of metal oxygen bonds leaves the Al–O network structure nearly unperturbed (isomorphous substitution). Contrary to the $\text{Al}_2\text{FeO}_4^+$ ($n = 1$), valence isomerism is not observed, which is attributed to a smaller M:O ratio ($M = \text{Al, Fe}$) and consequently decreasing electron affinities with increasing cluster size.

Introduction

Research on aluminium oxide has attracted considerable attention because of its versatile use, for example as photocatalyst, catalyst support,¹ coating,² or abrasive material, and in nanosensors.³ The desired characteristics of alumina for the different applications are based on its micro- and macroscopic structure as well as its electronic properties. It is, however, well known that alumina is rarely free of impurities, such as Fe^{3+} , Cr^{3+} , or Ti^{4+} ions, replacing Al^{3+} centers.⁴ With up to 9 wt%,

iron is the most common foreign ion in Al_2O_3 . Those impurities change the microscopic structure of the oxide and its macroscopic properties *e.g.* optical behavior, magnetism and catalytic activity.^{5–12}

Detailed experimental information about the change in local lattice structure due to incorporated Fe ions in solid Al_2O_3 is scarce. While changes of the electronic structure can be probed *via* UV/Vis adsorption,^{6,11,12} Mössbauer^{13,14} or electron paramagnetic resonance (EPR)¹⁵ spectroscopy, modern X-ray methods such as extended X-ray absorption fine structure (EXAFS) and X-ray absorption near edge structure (XANES) are necessary to determine coordination environments and bond distances of the metal centers. Kou and co-workers,¹⁶ as well as Tai and co-workers,¹⁷ report an average Fe–O bond distance of 198–202 pm for Fe-doped $\gamma\text{-Al}_2\text{O}_3$ and a coordination number (CN) of 4.9 for iron, indicating a mixture of tetrahedral and octahedral coordination. Grunwaldt and co-workers¹⁸ find a much larger value for the mean bond distance (219 pm) but almost the same CN (4.7). For the system Fe/ $\alpha\text{-Al}_2\text{O}_3$ Gaudry and co-workers¹⁹ give an average Fe–O bond distance of 198 pm and assume octahedral oxygen coordination. All authors emphasize the fact that two different oxygen coordination shells around iron exist (distorted coordination), resulting in shorter (~ 192 pm) and longer (> 205 pm) Fe–O bonds.

It is experimentally well known that the introduced $\text{Fe}^{3+}(\text{d}^5)$ ions are in a high-spin ($S = 5/2$) state.¹⁵ Solid state calculations showed that the iron impurity is most stable in the layers close to, or at the surface.^{20,21} Substituting an Al^{3+} with an Fe^{3+} ion in the topmost layer of the $\gamma\text{-Al}_2\text{O}_3$ (110) surface results in an energy gain of about 50 kJ mol^{-1} .²⁰

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† Electronic supplementary information (ESI) available: The PDF file contains the calculated IR spectra of several low-energy isomers for $(\text{Al}_2\text{O}_3)_{1-5}\text{FeO}^+$ in comparison with their respective IRPD spectrum (Fig. S1–S5), a Table with relative energies of all studied isomers ($M = \text{Fe, Al}$) obtained with different density functionals (Table S1), a table listing the experimentally observed IR bands for $(\text{Al}_2\text{O}_3)_{2-5}\text{FeO}^+$ (Table S2), and a figure showing relevant bond distances of the assigned isomers (Fig. S6). The ZIP compressed folder contains XYZ coordinate files for all calculated isomers. Each of these files also comprises the total electronic and zero point vibrational energy of the respective molecule. See DOI: <https://doi.org/10.1039/d2cp02938c>

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While for samples of doped bulk alumina the distribution of Fe impurities cannot be controlled and local structure information for the impurity is difficult to obtain, gas-phase studies on isolated metal oxide clusters of well-defined compositions provide the possibility to study the impurity site in isolation. In particular, the combination of cryogenic ion trap vibrational spectroscopy^{22,23} with density functional theory (DFT) is a reliable tool for the determination of the structure and the electronic properties of isolated polynuclear metal oxide clusters.^{23–26}

In the present study, we investigate how an Fe impurity influences the structure and magnetic properties of Al oxide clusters. While mass-selective experiments are difficult for neutral clusters, they can be readily performed on cluster ions. Therefore, we have previously used the $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$ series of cluster cations as model systems for alumina.²⁷ These clusters still feature a closed-shell electronic structure and exclusively trivalent aluminium and divalent oxygen ions²⁷ – just as bulk alumina. With increasing n , also the clusters' composition approaches the one of Al_2O_3 . Hence, it is expected that the clusters $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$ represent suitable model systems for the local structure of alumina in general and the distorted surface layers of Al_2O_3 in particular.

Here, we adopt the $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ series as model systems for Fe-doped Al_2O_3 . Substitution of one of the Al^{3+} ions with an Fe^{3+} ion converts the $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$ series to a model for $\text{Fe}/\text{Al}_2\text{O}_3$. Note that the nomenclature $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ does not imply that an FeO^+ unit is added to Al_2O_3 , but rather that an Al^{3+} has been replaced with Fe^{3+} in the closed-shell systems $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$.

Our goal is to identify the preferred substitution site and coordination environment of the Fe atom in the clusters of increasing size ($n = 2–5$) and hence decreasing iron-to-aluminium ratio. Furthermore, we are interested in the local electronic structure of the dopant, including its spin and oxidation state. A previous study on the first member of the series ($n = 1$),²⁸ $(\text{Al}_2\text{O}_3)\text{FeO}^+ \equiv \text{Al}_2\text{FeO}_4^+$, revealed the formation of an unexpected structure different from that of Al_3O_4^+ as a consequence of valence isomerism: In an intramolecular redox reaction, iron is reduced to $+\text{II}$, accompanied by the formation of a terminal oxygen radical. Consequently, the question arises if such a change of oxidation state also occurs in the larger clusters.

Experimental methods

The infrared photodissociation (IRPD) experiments were carried out on a 10 K ion-trap tandem mass spectrometer²⁹ using the widely tuneable, intense IR radiation from the Fritz-Haber-Institute Free Electron Laser (FHI FEL).³⁰ In brief, $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ ions ($n = 2–5$) are produced in a pulsed laser vaporization source by focusing a frequency-doubled Nd:YAG laser (50 Hz, 10–15 mJ) onto a rotating Fe/Al (30/70 wt%) metal rod (Plasmaterials). The resulting plasma is quenched with a gas pulse of 0.5% O_2 seeded in He. Cluster ions are formed during expansion through a clustering channel downstream from the rod and pass through a skimmer.

The beam of ions is then collimated and thermalized close to room temperature in a He-gas filled radio frequency (RF) ion guide, mass-selected using a quadrupole mass filter, and focused into a cryogenically cooled RF ring-electrode ion-trap. The trap is continuously filled with He-buffer gas and held at a temperature of 12 K. Many collisions of the trapped ions with the buffer gas provide gentle cooling of the internal degrees of freedom close to the ambient temperature. At sufficiently low ion-trap temperatures, ion-He complexes are formed *via* three-body collisions.³¹

All ions are extracted from the ion trap and focused both temporally and spatially into the center of the extraction region of an orthogonally-mounted reflectron time-of-flight (TOF) tandem mass spectrometer. Here, the ions are irradiated with an attenuated, counter-propagating IR laser pulse produced by the FHI FEL (430–1200 cm^{-1} , 5 Hz), with a bandwidth of $\sim 0.5\%$ full width at half maximum (FWHM) and a pulse energy of 0.7–2.5 mJ. Parent, as well as photofragment ion yields, are monitored simultaneously as a function of the irradiation wavelength. IRPD scans are recorded by averaging over 100 TOF mass spectra per wavelength step (3 cm^{-1}). Typically, at least three scans are summed to obtain the final IRPD spectrum. The photodissociation cross-section σ_{IRPD} is determined as described previously.^{32,33}

Computational methods

All calculations were performed with the Turbomole program package V7.5.1.^{34–36} A global energy minimization for the clusters of the compositions $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ ($n = 2–5$) was carried out employing a genetic algorithm (GA).^{37,38} The GA stopped when the energy of the most stable isomer as well as the averaged energy of all unique structures per generation did not change anymore. Thereby, 1200 structures were created for $(\text{Al}_2\text{O}_3)_2\text{FeO}^+$ and 1700 for each of the $(\text{Al}_2\text{O}_3)_{3–5}\text{FeO}^+$ systems.

In the GA optimizations, the BP86^{39,40} exchange-correlation density functional was used together with a split valence basis set (def2-SVP⁴¹) for the sake of computational efficiency. The total electronic spin of all investigated systems in this step was fixed at $S = 5/2$, *i.e.* the high-spin state of the Fe atom. Thus, the spin unrestricted Kohn–Sham (UKS) formalism was required to handle the open-shell electronic structure. The resolution of the identity (RI) approximation was used to accelerate the Coulomb fitting in the global structure search of the largest cluster ($n = 5$).

The most stable unique isomers of each composition were re-optimized employing the TPSSh,⁴² the PBE0⁴³ and the B3LYP^{44–46} exchange-correlation functionals in conjunction with the def2-TZVP triple-zeta basis set.⁴⁷ Vibrational frequencies were calculated within the double-harmonic approximation. The resulting stick spectra were convolved with Gaussian functions of 15 cm^{-1} full width at half maximum (FWHM) to account for the width of the experimentally observed bands.

All vibrational frequencies were scaled to account for anharmonic effects and the systematic error of the used density functional concerning the force constants. The scaling factor of 1.0175 was obtained by maximizing the cosine similarity score (*vide infra*) of the simulated TPSSh spectrum for $\text{Al}_2\text{FeO}_4^+$ ($n = 1$) and the respective IRPD result. This factor is slightly different from the one used in the previous study (1.0132)²⁸ since a different basis set was used.

The influence of the He-tagging atoms was investigated recently for the cluster $\text{Al}_2\text{FeO}_4^+$ ($n = 1$) and found to be negligible.²⁸ It is, therefore, not further considered in this work.

As an objective measure for the agreement of experimental and theoretical spectra, the cosine similarity score S is used. S expresses the similarity between the two spectra. It is based on the two vectors **A** and **B**, which hold the intensity values of the experimental and predicted spectrum at the same grid of irradiation energies, respectively. The score is calculated using eqn (1), *i.e.* by the normalized dot product of **A** and **B**.²⁸

$$S = \cos(\theta) = \frac{\mathbf{A} \cdot \mathbf{B}}{\|\mathbf{A}\| \|\mathbf{B}\|} = \frac{\sum_{i=1}^n A_i B_i}{\sqrt{\sum_{i=1}^n A_i^2} \sqrt{\sum_{i=1}^n B_i^2}} \quad (1)$$

The score can vary from zero to unity, and S values closer to unity indicate greater similarity.

Results and discussion

In Fig. 1, the IRPD spectra of He-tagged $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ with $n = 2-5$ are compared with those previously reported for the He-tagged all-Al analogs $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$ in the spectral region from 430 to 1200 cm^{-1} (see Table S2, ESI† for experimental band positions).²⁷ The spectrum of $\text{Al}_2\text{FeO}_4^+$ ($n = 1$) has been published recently by Müller *et al.*²⁸ Based on our previous studies for monometallic Al- and Fe-oxide clusters, we group the observed IR absorptions into three characteristic spectral regions: (i) 1200–900 cm^{-1} , (ii) 900–600 cm^{-1} and (iii) 600–430 cm^{-1} .^{48–52} These regions cover (i) Al–O stretching modes involving the shortest (≤ 173 pm) Al–O bonds, (ii) Fe–O stretching modes together with medium-length (174–178 pm) Al–O bond stretches and (iii) ring breathing modes as well as stretching and bending modes involving the longest (> 180 pm) Al–O bonds.

For each n , the spectra with (left) and without (right) Fe-substitution exhibit similar features, in particular in the region (i), but there are also noticeable differences in the number of peaks, their positions and intensities predominantly in regions (ii) and (iii). The degree of agreement between corresponding spectra increases with increasing n . This observation is supported by the cosine similarity score analysis, which yields S values of 0.49, 0.68, 0.85 and 0.86 for $n = 2-5$, respectively. As can be expected, a decreasing relative Fe content with

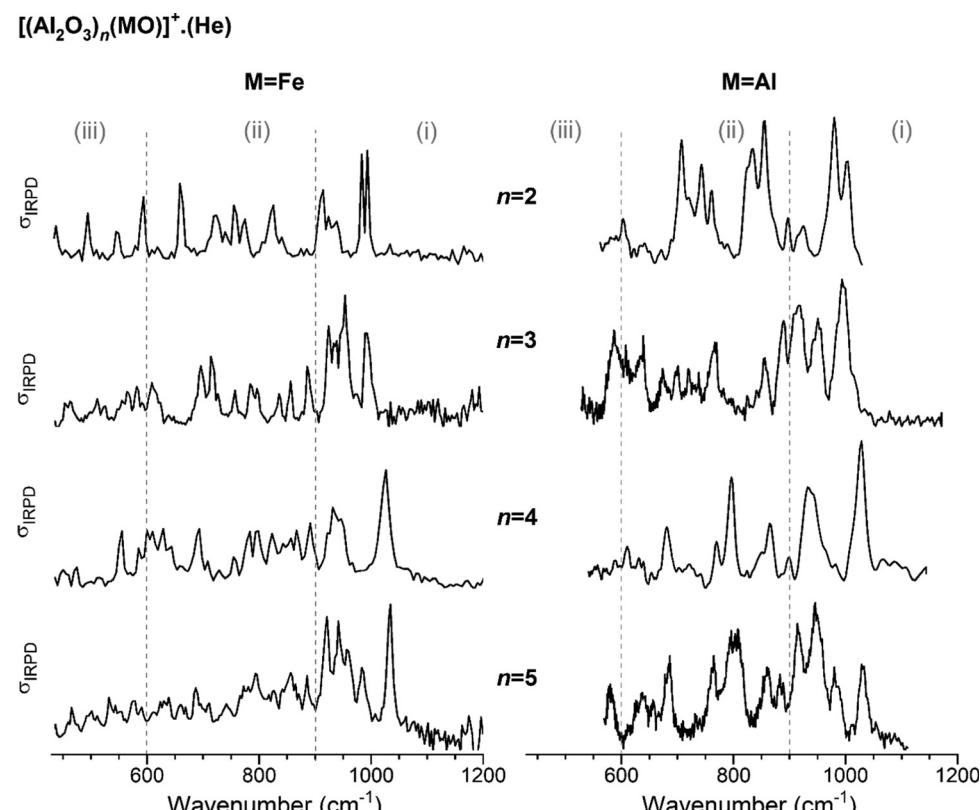


Fig. 1 Experimental IRPD spectra of He-tagged $(\text{Al}_2\text{O}_3)_n(\text{MO})^+$ with $n = 2-5$. Spectra for $M = \text{Fe}$ (present study, singly He-tagged clusters) are shown on the left-hand side and those for $M = \text{Al}$ (from ref. 27, multiple He-tagging) on the right-hand side panel. See Table S2, ESI† for band positions and assignments. Three characteristic spectral regions are indicated by the dashed lines: (i) 1200–900, (ii) 900–600, and (iii) $< 600 \text{ cm}^{-1}$.

increasing cluster size leads to decreasing distortion of the alumina framework and hence more similar IRPD spectra.

The most intense absorption bands of the $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ clusters are seen in the region (i), and the highest-energy transition is found in the spectrum of the largest cluster, $n = 5$, at 1034 cm^{-1} . Spectra of the all-Al species also show the most intense absorption in the region (i). Regions (ii) and (iii) are rich with IR signatures and the spectral congestion naturally increases with increasing cluster size for both types of clusters.

Fig. 2 gives an overview of isomers for the cluster compositions $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ resulting from global energy minimization using a genetic algorithm and DFT together with some isomers that were previously obtained for the pure aluminium oxide cations $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$.²⁷ All isomers are labelled with the notation nx_k , where n is the number of formal (Al_2O_3) units present in the cluster, $x = a, b, c$ indicates different framework motifs sorted by increasing relative energy, and k is an index to discriminate different Fe-substitution positions within a motif. To address pure Al-oxide systems, the right subscript "Al" is added. Note, the present labeling is different from that used in our original $(\text{Al}_2\text{O}_3)_{1-4}\text{AlO}^+$ study.²⁷

Calculated harmonic IR spectra of the different isomers for each cluster size were compared with their corresponding experimental spectrum to identify the best matching one. Fig. 3 shows the finally assigned structures and their IR spectra; see Fig. S1–S5 in the ESI† for the spectra of all remaining isomers.

The global minimum-energy isomer predicted for $\text{Al}_4\text{FeO}_7^+$ ($n = 2$) is a sheet-like structure ($2\mathbf{a}_1$) with a four-fold coordinated Al atom and a bridging two-fold coordinated O atom. The Fe atom is three-fold coordinated. Fig. S2, ESI† depicts other calculated low-energy isomers and compares their calculated IR spectra with the experimental one. The isomers $2\mathbf{a}_2$ and $2\mathbf{a}_3$ share their structural motif with $2\mathbf{a}_1$ but the Fe atom is located at different metal sites. Both are higher in energy, 52 and 95 kJ mol^{-1} , respectively (Table S1, ESI†). In $2\mathbf{a}_3$ the Fe atom takes the four-fold coordinated position.

Isomer $2\mathbf{a}_1$ (C_1 , no symmetry) represents the lowest energy structure and its spectrum results in the highest similarity score (Fig. S2, ESI†). However, no straightforward explanation for the splitting of the observed peak at 993 cm^{-1} can be given. A possible origin might be a Fermi resonance between the intense fundamental predicted at 999 cm^{-1} , which mainly results from the motion of bridging two-fold coordinated O atoms ($\text{Al}-\mu_2\text{O}-\text{Al}$), and a combination band consisting of framework modes.

The highest-energy experimental peak (993 cm^{-1}) involves stretching of the shortest Al–O bonds (169–173 pm). Al–O stretching bands of four- and three-fold coordinated Al atoms ($\mu_4\text{Al}/\mu_3\text{Al}$) are assigned to the experimental peaks at 825 cm^{-1} (calc. 830 cm^{-1}) and 593 cm^{-1} (calc. 598 cm^{-1}), respectively. Fe–O stretching modes exclusively appear in the region (ii); the corresponding bond lengths are 177–208 pm. The Fe–O motions are all coupled with the stretching of Al–O bonds of

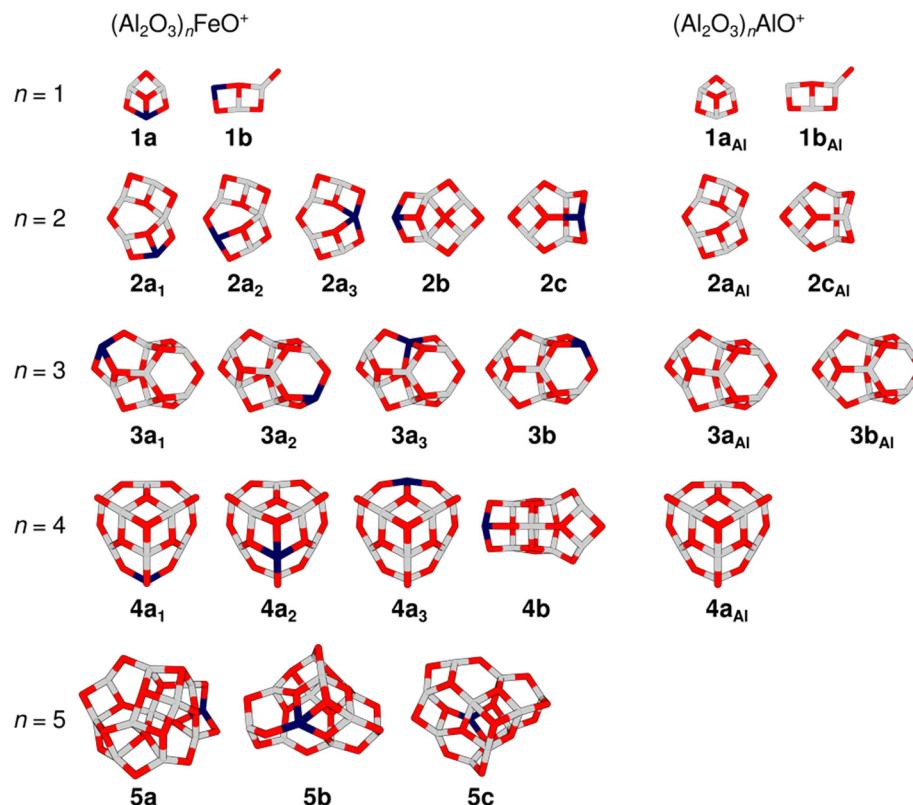


Fig. 2 Overview of TPSSh/def2-TZVP low-energy isomers of $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ (left-hand side) and $(\text{Al}_2\text{O}_3)_n\text{AlO}^+$ (right-hand side). Isomer labels are given below each structure (gray: aluminium, dark blue: iron, red: oxygen). Table S1 in the ESI† lists the relative energies of the isomers.

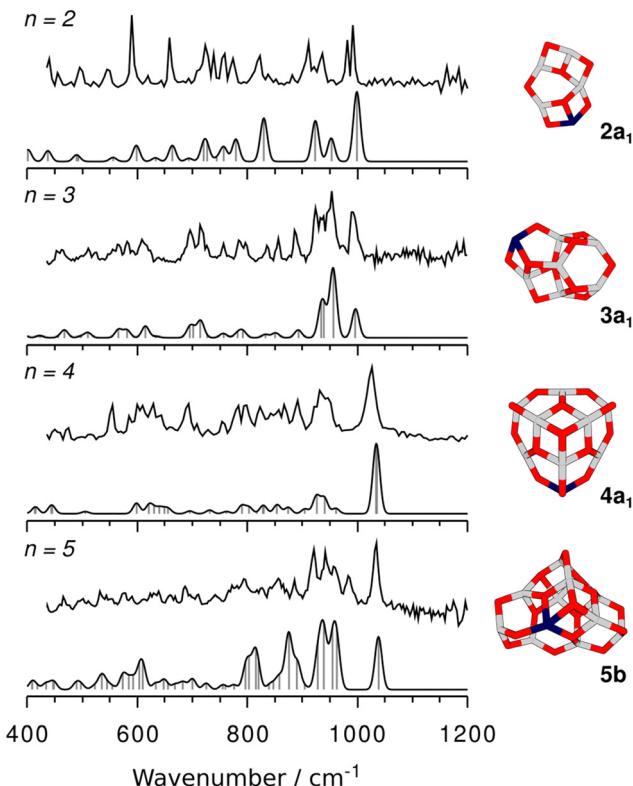


Fig. 3 Experimental IRPD spectra of He-tagged $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ (upper trace in each panel) compared with the scaled harmonic TPSSh/def2-TZVP spectra (lower trace in each panel, scaling factor: 1.0175) of the (un>tagged) assigned isomers. On the right-hand side the respective isomers are depicted (gray: aluminium, dark blue: iron, red: oxygen).

medium length (173–178 pm). Almost decoupled $\text{Fe}-\mu_2\text{O}-\text{Al}$ stretching modes are assigned to the transitions predicted at 664 and 632 cm^{-1} (experiment 659 and 619 cm^{-1}).

The IRPD spectrum of the all-Al analog Al_5O_7^+ (Fig. 1) contains contributions of two energetically close-lying isomers, $2\mathbf{a}_{\text{Al}}$ and $2\mathbf{c}_{\text{Al}}$ (ratio 1 : 3),²⁷ also shown in Fig. 2. While $2\mathbf{a}_{\text{Al}}$ is similar in structure to the sheet-like $2\mathbf{a}_1$, the analog of the cage-like structure $2\mathbf{c}_{\text{Al}}$ is $2\mathbf{c}$, which is 31 kJ mol^{-1} less stable than $2\mathbf{a}_1$ and is not observed at our experimental conditions. Indeed, a considerable difference in the experimental spectra of $\text{Al}_4\text{FeO}_7^+$ and Al_5O_7^+ (Fig. 1) can be seen mostly in the regions (i) and (ii), confirming these structural differences.

Proceeding to $\text{Al}_6\text{FeO}_{10}^+$, *i.e.* $n = 3$, the cage-like non-symmetric (C_1) structure $3\mathbf{a}_1$ (Fig. 3) is the most stable isomer and the origin of the measured IRPD spectrum. It consists of six- and four-membered rings, similar to the analogous $3\mathbf{a}_{\text{Al}}$ structure of $\text{Al}_7\text{O}_{10}^+$. The Fe atom is again three-fold coordinated with bond lengths between 178 and 206 pm.

The similarity of the IRPD spectra of $\text{Al}_6\text{FeO}_{10}^+$ and $\text{Al}_7\text{O}_{10}^+$ (*cf.* Fig. 1) supports the assumption of their identical structural motifs. While the spectrum of the pure alumina cluster was assigned to a 3 : 1 mixture of $3\mathbf{a}_{\text{Al}}$ and $3\mathbf{b}_{\text{Al}}$, most of the peaks observed for $\text{Al}_6\text{FeO}_{10}^+$ can be assigned to a single isomer. This is the lowest energy isomer $3\mathbf{a}_1$ and its simulated IR spectrum also yields the highest cosine similarity score $S = 0.84$ (Fig. S3,

ESI[†]). Unlike the IRPD spectra of all the other cluster sizes, the highest-energy peak for $n = 3$ is not the most intense one. This peak (992 cm^{-1}) is assigned to the stretching mode involving the shortest (170–173 pm) $\mu_3\text{Al}-\text{O}$ bonds from the outer rim of the cluster (calc.: 996 cm^{-1}). The most intense experimental band at 953 cm^{-1} results from the combination of stretching motions of $\mu_4\text{Al}-\text{O}$ units (calc.: 956 cm^{-1}). The $\mu_4\text{Al}-\text{O}$ bonds are shorter (174–175 pm) than those in $2\mathbf{a}_1$ and the corresponding IRPD band is blue-shifted compared to the spectrum of the $n = 2$ cluster and appears in the region (i). Region (ii) is rich with IR signatures from multiple Al–O and Fe–O stretching modes. $\text{Fe}-\mu_2\text{O}$ stretching motions are observed at 714 cm^{-1} (calc.: 715 cm^{-1}).

The global minimum-energy structure for $\text{Al}_8\text{FeO}_{13}^+$ is the conical structure $4\mathbf{a}_1$ (Fig. 3). Compared to its all-Al analog, its symmetry is reduced from C_{3v} to C_s upon substitution of a three-fold coordinated Al with an Fe atom in the outer rim of $4\mathbf{a}_{\text{Al}}$. The structure consists of a tip that is similar to the structure of Al_3O_4^+ ,²⁷ which is linked to a 12-membered Al_5FeO_6 ring by three bridging $\mu_2\text{O}$ atoms. Iron substitution at the other two unique metal positions leads to the isomers $4\mathbf{a}_2$ and $4\mathbf{a}_3$, which are 13 kJ mol^{-1} and 20 kJ mol^{-1} less favorable, respectively. While $4\mathbf{a}_1$ and $4\mathbf{a}_3$ feature three-fold coordinated Fe centers, the transition metal is bound to four O atoms in $4\mathbf{a}_2$. The simulated spectra of the three isomers $4\mathbf{a}_{1-3}$ yield very similar S values (Fig. S4, ESI[†]). However, all used DFT methods agree that $4\mathbf{a}_1$ is the most stable structure (Table S1, ESI[†]) and, hence, it is assigned to $\text{Al}_8\text{FeO}_{13}^+$.

Similar to the smaller clusters, the shortest Al–O bonds (169–172 pm) are found in the 12-membered ring forming the outer rim. The associated $\text{Al}-\mu_2\text{O}$ stretching bands are observed with the broad absorption at the highest energy (1026 cm^{-1}) having the highest intensity. The simulated spectrum reveals two very close-lying features at 1031 and 1033 cm^{-1} , which could contribute to the larger width of the experimental peak. All three Fe–O bonds have almost identical lengths (183 pm) and result in stretching bands, which can be found in region (iii).

In comparison to the spectrum of $\text{Al}_8\text{FeO}_{13}^+$, the $\text{Al}_9\text{O}_{13}^+$ spectrum shows fewer features, in line with its higher symmetry (C_s vs. C_{3v}). However, the features in the region (i) remain similar due to the identical outer ring structure of these isomers. The influence of the Fe atom on the IRPD spectrum and hence on the isomer-specific features manifests in the region 550–900 cm^{-1} due to slight changes in the framework introduced by the longer Fe–O bonds.

For the largest cluster, $\text{Al}_{10}\text{FeO}_{16}^+$, the assignment of the IRPD spectrum to a specific isomer is difficult. The most stable structure at the TPSSh level is $5\mathbf{a}$. Isomers $5\mathbf{b}$ and $5\mathbf{c}$ are, however, only 10 and 16 kJ mol^{-1} less favorable (Table S1, ESI[†]), *i.e.* within the uncertainty range of the applied functionals. The relative stabilities of the three isomers change when B3LYP and PBE0 instead of TPSSh are used (Table S1, ESI[†]). Both methods favor $5\mathbf{b}$, suggesting that the energetic ordering predicted by TPSSh might not be reliable in this particular case. Hence, it can be assumed that $5\mathbf{b}$ is more stable and therefore more abundant in the experiment.

All three isomers have irregular cage structures (Fig. 2) consisting of six- and four-membered rings. In **5a** and **5c**, the Fe atom is three-fold coordinated, whereas it is four-fold coordinated in **5b**. Compared to the IRPD spectrum, **5a** reaches the lowest similarity score. In contrast, **5b** and **5c** result in much better, almost identical spectral agreements (Fig. S5, ESI[†]). However, neither the spectrum of **5b** nor of **5c** alone reproduces all observed IRPD bands, indicating substantial contribution from at least one other low-energy isomer. The characteristic feature at 983 cm⁻¹, for example, could be explained by a contribution of **5a**. Still, taking into account, that none of the used density functionals predicts **5c** to be the most stable isomer, and that there is high spectral similarity for **5b** (*S* = 0.79), the latter is assumed the primarily present isomer of Al₁₀FeO₁₆⁺.

The most intense band in the IRPD spectrum of Al₁₀FeO₁₆⁺ at 1034 cm⁻¹ is not only the highest-energy absorption of this cluster but also among the series of mixed and all-Al oxide systems reported in this paper (Fig. 1). It is, just as the other features in region (i), associated with the stretching motions of the shortest bridging μ₂O-Al bonds (167–174 pm). At this cluster size, it becomes difficult to identify the IR-signature of the Fe center, because the corresponding stretching modes are strongly coupled with other vibrational modes of the system. The Fe-O bond lengths range from 179 to 202 pm.

Based on the assignments made, the structural influence of Fe-doping on small alumina clusters can be discussed. Table 1 lists the average Al-O bond lengths for (Al₂O₃)_nAlO⁺ and (Al₂O₃)_nFeO⁺ as well as all particular Fe-O bond lengths of the assigned isomers. Moreover, the (average) coordination numbers (CN) of the Al and Fe sites are given. Aluminium tends to form shorter bonds with oxygen than iron. The average length is about 176 pm, for both the pure and the substituted clusters. The only exception seems to be Al₆MO₁₀⁺ (*n* = 3, *M* = Al, Fe), where an averaged value of 180 pm arises. Indeed, there are some much longer Al-O bonds (193–208 pm and 188–200 pm for the undoped and doped cluster, respectively, *cf.* Fig. S6, ESI[†]) in the corresponding isomer, which increase the average value and the standard deviation. Most of them are associated

with the central, four-fold coordinated O atom. Its higher coordination number might be the reason for the weaker interaction with the surrounding Al atoms and, therefore, for the longer bond distances. This assumption is supported by the longer mean Al-O bonds in solid state α-Al₂O₃ (Table 1: >186 pm), where oxygen is also four-fold coordinated.

The iron oxygen bonds have a length of at least 177 pm and can be as long as 208 pm. In most clusters, there are one or two shorter (<184 pm) and one longer (>194 pm) Fe-O bonds. This agrees well with the experimental observation of shorter and longer Fe-O bonds in solid Fe-doped Al₂O₃ (Table 1). However, the absolute values slightly deviate: both the Al-O and the Fe-O bonds in the crystal lattice are longer than those of the gas phase clusters. As mentioned above, this is a consequence of the higher CNs of both metal and oxygen atoms in the solid state and the resulting weaker atom-to-atom interactions.

With decreasing metal-to-oxygen ratio (*M*:O), the (mean) CN of Al, as well as Fe, increases, reaching a value of (or close to) four for Al₁₀FeO₁₆⁺. In corundum or hematite each metal site has octahedral oxygen coordination. Hence, larger clusters with *M*:O approaching 2:3 will also tend towards a CN of six. It is also interesting to see that in Fe/γ-Al₂O₃ the Fe atoms are surrounded by two oxygen coordination shells, responsible for the longer and shorter Fe-O bonds.^{16,18} Each of those shells results in a (partial) CN of about 2.4 for iron, which is in turn comparable to the valency of the Fe in the small clusters (3.0).

Analysis of the structures of (Al₂O₃)_nFeO⁺ (*n* = 1–5) reveals that Al prefers trigonal planar or tetrahedral O atom coordination while Fe seems more flexible and adapts also to two-fold coordinated environments.²⁸ These observations are in line with the results for Fe-doped nanocrystals or nanoparticles,¹² in which dopant Fe atoms preferably replace Al atoms of the Al₂O₃ network mostly in the surface layer²¹ without inducing major distortions or even structural changes.⁵³ While structural deviations from the corresponding all-Al clusters are evident for the clusters with larger Fe:Al ratios (*n* = 1–3), starting from Al₈MO₁₃⁺ the cluster framework is virtually identical for *M* = Fe and Al, proving the suitability of iron as a dopant for Al₂O₃.

Table 1 Structural parameters (bond distances in pm, coordination numbers [CN], metal-to-oxygen ratio [M:O]) and electron affinities (in eV) of the assigned isomers for (Al₂O₃)_nFeO⁺ obtained with TPSSh/def2-TZVP. For comparison, experimental reference values are given for Fe-doped solid Al₂O₃

	M:O	Bond distances [pm]			CN		
		Al-O ^{ab}	Al-O ^{ac}	Fe-O	Al ^d	Fe	EA _{vert} ^e
1b	0.750	176 ± 8	175 ± 7	183, 194	3.0	2.0	7.31 ^f
2a₁	0.714	176 ± 7	176 ± 6	177, 181, 208	3.3	3.0	6.65
3a₁	0.700	180 ± 12	180 ± 11	178, 179, 205	3.7	3.0	6.34
4a₁	0.692	175 ± 6	175 ± 6	183, 183, 183	3.4	3.0	5.68
5b	0.688	180 ± 10	178 ± 7	179, 191, 200, 202	3.6	4.0	6.00
Fe/α-Al ₂ O ₃ ^g	0.667	186, 197		190, 205			
Fe/γ-Al ₂ O ₃ ^h	0.667			191, 205		4.9	
Fe/γ-Al ₂ O ₃ ⁱ	0.667			192, 245		4.7	
Fe/γ-Al ₂ O ₃ ^j	0.667			194, 209		6.0	

^a Average Al-O bond distances with uncertainty of one standard deviation. ^b Reference values for all-Al clusters, (Al₂O₃)_nAlO⁺. ^c Fe-doped cluster, (Al₂O₃)_nFeO⁺. ^d Averaged over all Al atoms in the respective Fe-doped cluster. ^e Vertical electron affinity of Fe-doped clusters. ^f Electron affinity for **1a**. ^g Taken from ref. 19. ^h Taken from ref. 16. ⁱ Taken from ref. 18. ^j Taken from ref. 17.

Regarding spin multiplicity and oxidation state of the iron centers, it turns out that a Fe(III) high-spin d⁵ configuration (sextet, $S = 5/2$) is always the most stable situation. The presence of a high-spin Fe(III)-center in $(\text{Al}_2\text{O}_3)_{2-5}\text{FeO}^+$ is in agreement with observations for solid Fe/Al₂O₃, where EPR and Mössbauer spectroscopy suggests Fe(III) high-spin sites.¹⁵

Table S1 (ESI†) lists the relative stabilities of different spin isomers calculated with several density functionals (TPSSh, B3LYP, PBE0), showing that the preference for high-spin Fe(III) is independent of the choice of functional. At the TPSSh level, the quartets ($S = 3/2$) and doublets ($S = 1/2$) are 18–65 kJ mol⁻¹ and 75–119 kJ mol⁻¹ less stable than the sextets, respectively. The orbital analysis (see $\langle S^2 \rangle$ values, Table S1, and spin density plots, Fig. S7, ESI†) reveals that the quartet states feature three unpaired electrons in the d states ($\langle S^2 \rangle$ close to 3.75). In the doublets, however, there are also three unpaired electrons, two with spin up, and one with spin down, resulting in $\langle S^2 \rangle$ values close to 1.75 instead of 0.75 for only a single unpaired electron. It can be concluded that a strong multireference character of the transition metal sites in the larger clusters ($n = 2-5$) is absent for the sextets and quartets, so that DFT is a suitable method to describe them. The doublets, however, should be handled with more sophisticated wavefunction-based methods.

Contrary to $\text{Al}_2\text{FeO}_4^+$ ($n = 1$), the clusters with $n = 2-5$ are not subject to an intramolecular redox reaction. Hence, terminal oxygen radicals are absent. A possible reason for this can be the M:O ratio, which, for $n = 2-5$, is smaller than 0.75 (value for magnetite), tending towards 0.667 (value for hematite) with increasing cluster size. In an infinitely large cluster the latter ratio is approached and all metal sites would formally be in oxidation state +III. For the magnetite value (0.75), however, there needs to be a mixture of sites with +III and +II. Smaller M:O values intrinsically require higher formal oxidation states of the metal sites.

The decomposition of the isomerization energy from the isomer with Fe^{3+} (d⁵)/O²⁻ to the one with Fe^{2+} (d⁶)/O^{•-} into electron affinities (EA) and the isomerization energy between the neutral species was helpful in the case of $\text{Al}_2\text{FeO}_4^+$.²⁸ In this decomposition, the electron affinity of the Fe(III)-containing isomer is the driving force for the valence isomerism since both the Fe(II)-O bond cleavage and the re-ionization (O radical formation) consume energy. Table 1 contains the calculated EA values for the clusters with $n = 1-5$. They are much smaller (<6.7 eV) for $(\text{Al}_2\text{O}_3)_{2-5}\text{FeO}^+$ than for $\text{Al}_2\text{FeO}_4^+$ (7.3 eV). As described above, the smaller M:O ratio of the larger clusters favors, *i.e.* stabilizes the higher +III oxidation state of the Fe site and prevents the oxygen radical formation.

Conclusions

In agreement with the results for Fe-doped Al₂O₃ nanoparticles¹² and solid corundum, the substitution of a single Al atom with an Fe atom has little influence on the structure of the larger clusters of the homolog series $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ ($n = 1-5$). Only the first two

members ($n = 1, 2$) significantly change their structure, as can be concordantly seen from the experiment (spectra) and theory (structure prediction). A slightly different bonding environment is observed between Fe and O, compared to the Al-O bonds. The former bonds are longer and high-spin Fe(III) is found in three-fold coordination, while Al sites with four-fold coordination increase the cluster's stability. The Fe atom preferentially occupies lower-coordination sites at the rim of the clusters. With increasing cluster size ($n > 1$) the M:O ratio approaches the ideal value of hematite (0.667), leading to decreasing electron affinities and, as a consequence, the absence of $\text{Fe}^{3+}/\text{O}^{2-} \leftrightarrow \text{Fe}^{2+}/\text{O}^{\bullet-}$ valence isomerism that is present in $\text{Al}_2\text{FeO}_4^+$ ($n = 1$).

This study reveals an intimate relationship between structural and electronic properties of the investigated $(\text{Al}_2\text{O}_3)_n\text{FeO}^+$ gas phase clusters and both solid and nanoparticle-like Fe-doped Al₂O₃. This observation confirms their use as model systems⁵⁴ for further studies, *e.g.* reactivity studies with H₂, H₂O, CO, and CO₂, the educts and byproducts of the Fischer-Tropsch process, which uses calcinated Fe/Al₂O₃ as a catalyst. Furthermore, together with Fe atoms, strong local magnetic moments (d⁵ high-spin) are introduced in the alumina clusters enabling a possible application in spin-based quantum devices. An investigation of alumina models with two Fe-substituted metal sites and their resulting spin couplings could elucidate the suitability of such systems.

Author contributions

S. D. planned and performed the experimental work, including evaluation of the data using the cosine similarity score. M. J. assisted during the beam-time measurements. F. M. did the electronic structure calculations, assisted by J. B. S. for the systems $n = 3$ and $n = 5$. S. D. and F. M. wrote the manuscript and prepared the figures. F. B. supervised F. M., J. S. and K. R. A. designed the study, coordinated the project, supervised the research, contributed to the data interpretation and revised the manuscript.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 M. Trueba and S. P. Trasatti, *Eur. J. Inorg. Chem.*, 2005, 3393–3403.

2 G. V. Franks and Y. Gan, *J. Am. Ceram. Soc.*, 2007, **90**, 3373–3388.

3 J. Álvarez, C. Serrano, D. Hill and J. Martínez-Pastor, *Opt. Lett.*, 2013, **38**, 1058–1060.

4 A. W. Deer, R. A. Howie and J. Zussman, *An Introduction to the Rock-Forming Minerals*, The Mineralogical Society, London, 3rd edn, 2013.

5 N. Kristianpoller, A. Rehavi, A. Shmilevich, D. Weiss and R. Chen, *Nucl. Instrum. Methods Phys. Res.*, 1998, **411**, 343–346.

6 T. Monteiro, C. Boemare, M. J. Soares, E. Alves, C. Marques, C. McHargue, L. C. Ononye and L. F. Allard, *Nucl. Instrum. Methods Phys. Res.*, 2002, **491**, 638–643.

7 E. Gaudry, D. Cabaret, P. Sainctavit, C. Brouder, F. Mauri, J. Goulon and A. Rogalev, *J. Phys.: Condens. Matter*, 2005, **17**, 5467–5480.

8 V. B. Mikhailik, P. C. F. Di Stefano, S. Henry, H. Kraus, A. Lynch, V. Tsybulskyi and M. A. Verdier, *J. Appl. Phys.*, 2011, **109**, 053116.

9 G. Rani and P. D. Sahare, *Adv. Powder Technol.*, 2014, **25**, 767–772.

10 J. A. Loiland, M. J. Wulfers, N. S. Marinkovic and R. F. Lobo, *Catal. Sci. Technol.*, 2016, **6**, 5267–5279.

11 Z. K. Heiba, M. B. Mohamed, A. M. Wahba and N. G. Imam, *J. Electron. Mater.*, 2018, **47**, 711–720.

12 M. Farahmandjou, A. Khodadadi and M. Yaghoubi, *J. Supercond. Nov. Magn.*, 2020, **33**, 3425–3432.

13 H.-J. Wan, B.-S. Wu, C.-H. Zhang, H.-W. Xiang, Y.-W. Li, B.-F. Xu and F. Yi, *Catal. Commun.*, 2007, **8**, 1538–1545.

14 V. G. d Resende, A. Cordier, E. D. Grave, C. Laurent, S. G. Eeckhout, G. Giuli, A. Peigney, G. M. d Costa and R. E. Vandenberghe, *J. Phys. Chem. C*, 2008, **112**, 16256–16263.

15 A. Priem, P. J. M. van Bentum, W. R. Hagen and E. J. Reijerse, *Appl. Magn. Reson.*, 2001, **21**, 535–548.

16 Y. Kou, H.-l Wang, J.-z Niu and W.-j Ji, *J. Phys. Chem.*, 1996, **100**, 2330–2333.

17 A. Tomita, T. Miki, T. Tango, T. Murakami, H. Nakagawa and Y. Tai, *Chem. Phys. Chem.*, 2015, **16**, 2015–2020.

18 A. Boubnov, H. Lichtenberg, S. Mangold and J. D. Grunwaldt, *J. Phys. Conf. Ser.*, 2013, **430**, 012054.

19 E. Gaudry, A. Kiratisin, P. Sainctavit, C. Brouder, F. Mauri, A. Ramos, A. Rogalev and J. Goulon, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **67**, 094108.

20 G. Feng, C.-F. Huo, Y.-W. Li, J. Wang and H. Jiao, *Chem. Phys. Lett.*, 2011, **510**, 224–227.

21 M. Jiang, B. Nie, Z. Yang, M. Ni and Z. Chen, *J. Nucl. Mater.*, 2020, **533**, 152102.

22 K. R. Asmis, M. Brümmer, C. Kaposta, G. Santambrogio, G. von Helden, G. Meijer, K. Rademann and L. Wöste, *Phys. Chem. Chem. Phys.*, 2002, **4**, 1101–1104.

23 K. R. Asmis, *Phys. Chem. Chem. Phys.*, 2012, **14**, 9270–9281.

24 K. R. Asmis and J. Sauer, *Mass Spectrom. Rev.*, 2007, **26**, 542–562.

25 K. R. Asmis and J. Sauer, *Mass Spectrom. Rev.*, 2008, **27**, 205.

26 H. Schwarz and K. R. Asmis, *Chem. – Eur. J.*, 2019, **25**, 2112–2126.

27 G. Santambrogio, E. Janssens, S. Li, T. Siebert, G. Meijer, K. R. Asmis, J. Döbler, M. Sierka and J. Sauer, *J. Am. Chem. Soc.*, 2008, **130**, 15143–15149.

28 F. Müller, J. B. Stückrath, F. A. Bischoff, L. Gagliardi, J. Sauer, S. Debnath, M. Jorewitz and K. R. Asmis, *J. Am. Chem. Soc.*, 2020, **142**, 18050–18059.

29 D. J. Goebbert, G. Meijer and K. R. Asmis, *AIP Conf. Proc.*, 2009, **1104**, 22–29.

30 W. Schöllkopf, S. Gewinner, H. Junkes, A. Paarmann, G. von Helden, H. P. Bluem and A. M. M. Todd, *The new IR and THz FEL facility at the Fritz Haber Institute in Berlin*, SPIE, 2015.

31 M. Brümmer, C. Kaposta, G. Santambrogio and K. R. Asmis, *J. Chem. Phys.*, 2003, **119**, 12700–12703.

32 N. Heine and K. R. Asmis, *Int. Rev. Phys. Chem.*, 2015, **34**, 1–34.

33 N. Heine and K. R. Asmis, *Int. Rev. Phys. Chem.*, 2016, **35**, 507.

34 R. Ahlrichs, M. K. Armbruster, R. A. Bachorz, H. Bahmann, A. Baldes, M. Bär, H.-P. Baron, R. Bauernschmitt, F. A. Bischoff and S. Böcker *et al.*, TURBOMOLE (7.5.1), TURBOMOLE GmbH, 2021, <https://www.turbomole.org>.

35 O. Treutler and R. Ahlrichs, *J. Chem. Phys.*, 1995, **102**, 346–354.

36 K. Eichkorn, F. Weigend, O. Treutler and R. Ahlrichs, *Theor. Chem. Acc.*, 1997, **97**, 119–124.

37 M. Sierka, J. Döbler, J. Sauer, G. Santambrogio, M. Brümmer, L. Wöste, E. Janssens, G. Meijer and K. R. Asmis, *Angew. Chem., Int. Ed.*, 2007, **46**, 3372–3375.

38 D. M. Deaven and K. M. Ho, *Phys. Rev. Lett.*, 1995, **75**, 288–291.

39 J. P. Perdew, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 8822–8824.

40 A. D. Becke, *Phys. Rev. A: At., Mol., Opt. Phys.*, 1988, **38**, 3098–3100.

41 A. Schäfer, H. Horn and R. Ahlrichs, *J. Chem. Phys.*, 1992, **97**, 2571–2577.

42 V. N. Staroverov, G. E. Scuseria, J. Tao and J. P. Perdew, *J. Chem. Phys.*, 2003, **119**, 12129–12137.

43 C. Adamo and V. Barone, *J. Chem. Phys.*, 1999, **110**, 6158–6170.

44 S. H. Vosko, L. Wilk and M. Nusair, *Can. J. Phys.*, 1980, **58**, 1200–1211.

45 C. Lee, W. Yang and R. G. Parr, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1988, **37**, 785–789.

46 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 5648–5652.

47 F. Weigend and R. Ahlrichs, *Phys. Chem. Chem. Phys.*, 2005, **7**, 3297–3305.

48 P. Gruene, J. T. Lyon, G. Meijer, P. Lievens and A. Fielicke, *Ideas in Chemistry and Molecular Sciences*, 2010, pp. 183–206, DOI: [10.1002/9783527630530.ch7](https://doi.org/10.1002/9783527630530.ch7).

49 X. Song, M. R. Fagiani, S. Gewinner, W. Schöllkopf, K. R. Asmis, F. A. Bischoff, F. Berger and J. Sauer, *J. Chem. Phys.*, 2016, **144**, 244305.

50 X. Song, M. R. Fagiani, S. Gewinner, W. Schöllkopf, K. R. Asmis, F. A. Bischoff, F. Berger and J. Sauer, *Chem. Phys. Chem.*, 2017, **18**, 868–872.

51 M. R. Fagiani, X. Song, S. Debnath, S. Gewinner, W. Schöllkopf, K. R. Asmis, F. A. Bischoff, F. Müller and J. Sauer, *J. Phys. Chem. Lett.*, 2017, **8**, 1272–1277.

52 T. M. Maier, A. D. Boese, J. Sauer, T. Wende, M. Fagiani and K. R. Asmis, *J. Chem. Phys.*, 2014, **140**, 204315.

53 A. Khodadadi, M. Farahmandjou, M. Yaghoubi and A. R. Amani, *Int. J. Appl. Ceram. Technol.*, 2019, **16**, 718–726.

54 J. Sauer and H.-J. Freund, *Catal. Lett.*, 2015, **145**, 109–125.