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Probing the ionic structure of FLiNaK-ZrF_4 salt mixtures by solid-state NMR[†]

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In this study, by applying ^{19}F , ^{23}Na and ^7Li high-resolution NMR methods, the evolution of the $[\text{Zr}_x\text{F}_y]^{4x-y}$ local ionic structures in FLiNaK-ZrF_4 salt mixtures were elucidated. K_3ZrF_7 , Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystal phases were identified when the melt salts were being solidified. The distribution of these $[\text{Zr}_x\text{F}_y]^{4x-y}$ species was dependent on the content of ZrF_4 in FLiNaK eutectic salts. Moreover, K_3ZrF_7 phase transition from an orthorhombic lattice into a disordered cubic lattice was clarified, thereby causing dynamics of the coordinated F^- ions to be reduced and the well-ordered crystal lattices to be destroyed. These mentioned results provide a further insight into the Zr-F based ionic structure and the formation of the disordered Zr-F structure in ZrF_4 -based eutectic salts.

1 Introduction

Fluoride eutectic salts have been studied extensively as a coolant or a fuel salt for nuclear reactor systems due to their ability to act as an effective heat transfer fluid at high temperature and low pressure and based on high radiation flux.¹⁻⁴ Specific to the fuel salt, the concentration of free oxygen dianion O^{2-} is rigorously regulated to avoid UO_2 ⁵⁻⁷ being precipitated. As reported by Oak Ridge National Laboratory (ORNL), the zirconium tetrafluoride (ZrF_4) can act as the O^{2-} absorber additive in fuel salts for removing free O^{2-} and avoiding UO_2 being precipitated.⁸ However, ZrF_4 exerts a significant coordination effect with fluorine anions to form a series of $[\text{Zr}_x\text{F}_y]^{4x-y}$ anions, in which the Zr-F-Zr bridges bond chains or networks. In addition, the competition of F^- anions with other metal ions will drastically impact the salt properties (*e.g.*, the melt viscosity and thermo-conductivity). Accordingly, the structure study on ZrF_4 containing alkali fluorides is capable of presenting valuable information to support the novel fuel salt design, modeling and composition optimization.

To analyze the local structure and phase transformation of $[\text{Zr}_x\text{F}_y]^{4x-y}$ ions in AF-ZrF_4 systems ($\text{A} = \text{Li}^+, \text{Na}^+, \text{K}^+$), numerous experimental and theoretical studies have been conducted. According to Toth *et al.*,⁹ the effects of the ZrF_4 concentration on the local ionic structure in LiF-NaF-ZrF_4 eutectic salt were studied under Raman spectroscopy. Three zirconium-based species, *i.e.*, $[\text{ZrF}_6]^{2-}$, $[\text{ZrF}_7]^{3-}$ and $[\text{ZrF}_8]^{4-}$, coexist in the

melts. As reported by Dracopoulos *et al.*,¹⁰ the 6-fold and 7-fold coordinated ionic species could form a series of small chains in KF-ZrF_4 eutectic salt. With the F-Zr-F chains formed, the potential to form disordered network structures was illustrated. However, the specific evolution of the ionic structure and the crystal phase formation have been rarely investigated when complicated FLiNaK-ZrF_4 salts are being solidified, as impacted by the overlapped characteristic signals of the IR or Raman spectroscopy. As compared with other IR or Raman spectroscopy, high-resolution NMR method refers to a powerful method to investigate the local structure of ZrF_4 -based systems.¹¹⁻¹⁴ The high resolution of the NMR signals and the multiple detectable elements (^7Li , ^{19}F , ^{23}Na) are capable of presenting specific microstructure information of the zirconium ions in complex salt system. For instance, Pauvert *et al.*¹⁵ reported the structural evolution of the free fluoride ions and other Zr-F ions in LiF-ZrF_4 system. When ZrF_4 component tended to increase, the bridged Zr-F structure was certified to form Zr-F-Zr chains based on ^{19}F NMR method.

In this study, the coordination structure of $[\text{Zr}_x\text{F}_y]^{4x-y}$ in FLiNaK-ZrF_4 eutectic salts was investigated specifically with high resolution solid-state NMR method.¹⁶ ^7Li , ^{19}F and ^{23}Na NMR spectra were adopted to analyze the chemical environment of zirconium ions, as well as the evolution of the different ionic species. As indicated from the results, K_3ZrF_7 , Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystal phases were formed when the melt FLiNaK-ZrF_4 salts were being solidified. With the increase in the ZrF_4 concentration, K_3ZrF_7 phase transition from the crystalline phase into the disordered cubic was characterized, thereby suggesting that the dynamics of the coordinated F^- ions was reduced, and the well-ordered crystal lattices were destructed. This study helps clarify the Zr-F based ionic structure and the evolution of the $[\text{Zr}_x\text{F}_y]^{4x-y}$ species, which is critical to the

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solidification, energy storage and segregation of the ZrF_4 -based molten salts.

2 Experimental method

2.1 Samples preparation

The NaF (99.99%), KF (99.99%) and LiF (99.99%) were purchased from Aladdin and dehydrated by heating under vacuum at the temperature of 353 K for one week before use, and ZrF_4 (99.99%) with a cover wrapped by beeswax from Strem Chemicals, Inc without further purification. The highly-purified LiF-NaF-KF (46.5 : 11.5 : 42 mol%) eutectic salt was supplied by Shanghai Institute of Organic Chemistry, Chinese Academy of Sciences (inductively coupled plasma optical emission spectroscopy, ICP-OES results were shown in Table S1†), without further purification. Considering the eutectic point of 727 K, the $(\text{FLiNaK})_{\text{eut}}\text{-ZrF}_4$ eutectic solidification salts with various ZrF_4 compositions were prepared in a glove box under dried argon by artificially mixing suitable proportions between $(\text{FLiNaK})_{\text{eut}}$ salt and ZrF_4 at 923 K for keeping 4 h then that directly cool down to room temperature in the furnace. The related information that the number of kinds of ions and the ratio of $n(\text{F}^-)$ to $n(\text{Zr}^{4+})$ in FLiNaK-ZrF_4 ($0 \leq X_{\text{ZrF}_4} \leq 18.3$ mol%) systems were shown in Table S2.† The eutectic NaF-ZrF_4 (40.5 mol%) and NaF-ZrF_4 (20 mol%) salts were synthesized by the following: 1 g of salts were weighted by artificially mixing suitable proportions in a nickel crucible, then they were heated to 923 K and 1073 K respectively, and kept for 4 hours. After that, the temperature was directly cooled down to room temperature in the furnace. In order to help the discussion, the chemical shift of major compositions measured in ^{19}F MAS NMR was summarized in Table S3.†

2.2 Solid-state NMR experiments

All of the ^{19}F , ^{23}Na and ^7Li solid-state NMR experiments were performed on a Bruker Avance NEO 400 WB spectrometer with a magnetic field of 9.4 T, operating at frequencies of 376.61 MHz, 105.87 MHz and 155.55 MHz, respectively. The data was collected using a 3.2 mm double-resonance magic angle spinning (MAS) probe and a 15 kHz spinning rate at room temperature. In order to make quantitative experiments more effectively, the following conditions were set: the recycle delay (d1) were 5000 s, 100 s, 3000 s and the number of scans (ns) were 4, 16, 2 for ^{19}F , ^{23}Na and ^7Li solid-state NMR experiments respectively. The chemical shifts of ^{19}F , ^{23}Na and ^7Li were referenced using 1 M $\text{C}_2\text{H}_4\text{O}_2\text{F}_3\text{N}$ aqueous solution ($\delta = -74.5$ ppm), 1 M NaCl aqueous solution ($\delta = 0$ ppm) and 1 M LiCl aqueous solution ($\delta = 0$ ppm) at room temperature, respectively.

2.3 X-ray diffraction measurements

X-ray diffraction was performed at room temperature on a Bruker D8 Advance using $\text{Cu-K}\alpha$ (1.5406 \AA) radiation (40 kV, 20 mA). All samples were mounted on the same sample holder and scanned from $2\theta = 5^\circ$ to 90° at a speed of $15^\circ \text{ min}^{-1}$.

3 Results and discussion

3.1 Formation of the K_3ZrF_7 crystal phase in FLiNaK-ZrF_4 eutectic salt

The ionic structure of FLiNaK-ZrF_4 eutectic salts were firstly studied by comparing with FLiNaK . For the investigation of the local structure of the ions, ^{19}F , ^{23}Na and ^7Li solid-state MAS NMR was performed (Fig. 1). The characteristic signal of ZrF_4 could not be identified in ^{19}F NMR spectrum of FLiNaK-ZrF_4 (3.56 mol%) system (Fig. S1†),^{17,18} which demonstrated that no ZrF_4 crystal phase existed in the eutectic salt samples. Comparing to the pure FLiNaK eutectic solidification salt in which the signal of KF , LiF and NaF was located at -132.9 ppm, -205.4 ppm and -224.6 ppm respectively,¹⁹ a novel signal at -35.0 ppm was detected, which complied well with the reported result that has the characteristic signal at -34.8 ppm²⁰ and the XRD results of the authors (Fig. S2†). Moreover, the intensity of KF tended to decrease, which could be induced by the appearance of the new crystal phase. Given the mentioned analysis, the new crystal phase in FLiNaK-ZrF_4 eutectic salt was demonstrated as K_3ZrF_7 crystals. It was interesting to note that no novel signals were identified in the spectrum of ^{23}Na NMR and ^7Li NMR (Fig. 1b and c), thereby suggesting that NaF and LiF were not involved in the complexation with ZrF_4 in FLiNaK-ZrF_4 system. Moreover, no major change in the integration of ^{23}Na NMR and ^7Li NMR signal of NaF and LiF appeared (Fig. S3 and S4†) when the composition of ZrF_4 was changing from 0 to 13.6 mol%, indicating that NaF and LiF were not involved in the complexation with ZrF_4 in FLiNaK-ZrF_4 ($X_{\text{ZrF}_4} \leq 13.6$) systems.

To explore the formation of the new K_3ZrF_7 crystals in depth, FLiNaK-ZrF_4 salts exhibiting different ZrF_4 contents were synthesized. ^{19}F NMR was performed on the mentioned samples to elucidate the variation of different crystals (Fig. S5†), and the integration of the relative intensity of the mentioned crystals was plotted in Fig. 2a.

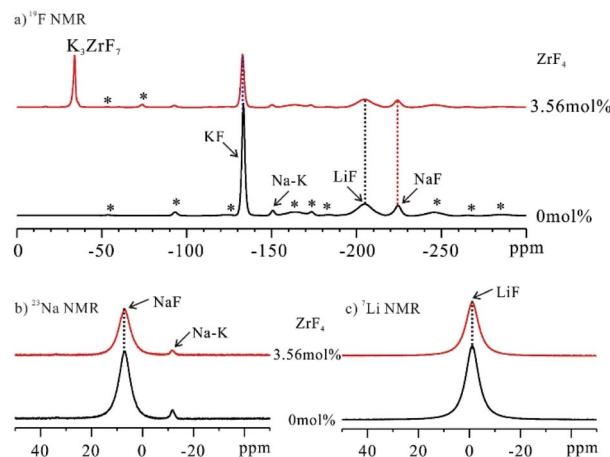


Fig. 1 ^{19}F (a), ^{23}Na (b) and ^7Li (c) solid-state MAS NMR spectra of FLiNaK eutectic salt (black lines) and FLiNaK-ZrF_4 (3.56 mol%) eutectic solidification salt (red lines) at ambient temperature. MAS spin rate in the mentioned experiments was set to 15 kHz. Spinning sidebands were marked with asterisks.



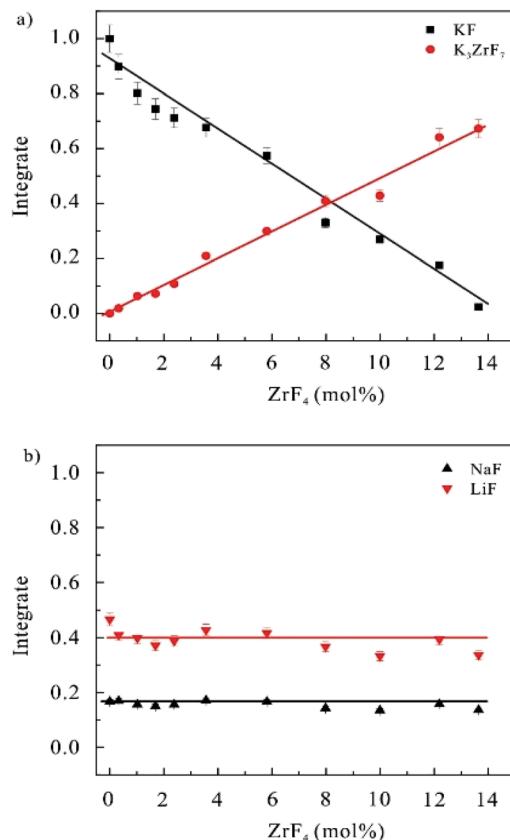


Fig. 2 Integrate of the ^{19}F NMR signals of different components in FLiNaK– ZrF_4 ($0 \leq X_{\text{ZrF}_4} \leq 13.6$ mol%) systems versus the molar fraction of ZrF_4 components.

As the ZrF_4 contents increased, the number of free fluoride ions F^- of KF crystal phase declined linearly, while the amount of K_3ZrF_7 tended to increase. Under the concentrations of ZrF_4 reaching 13.6 mol%, the signal of KF crystal phase at the -132.9 ppm almost disappeared, and the signal of K_3ZrF_7 phase at -35.0 ppm increased to the maximum. Thus, the added Zr^{4+} ions were coordinated with the free F^- ions, and the K_3ZrF_7 crystal phase were formed first when the molten salt was being solidified as impacted by the weaker bonding energy between K^+ and F^- ions than Na^+-F^- and Li^+-F^- ion pairs. Fig. 2b plots the evolving curves of the ^{19}F NMR signal intensity of NaF and LiF crystals. No variations of the integration were identified, which suggested that NaF and LiF crystals existed stably in FLiNaK– ZrF_4 ($0 \leq X_{\text{ZrF}_4} \leq 13.6$ mol%) salts and were not involved in the complexation with ZrF_4 . Accordingly, only zirconium-based species K_3ZrF_7 was proven to be formed in FLiNaK– ZrF_4 ($0 \leq X_{\text{ZrF}_4} \leq 13.6$ mol%) solidification salts.

3.2 Complexation between ZrF_4 and NaF

Given the mentioned analysis, ionic structure of Na^+ ions remained unchanged in FLiNaK– ZrF_4 salts with ZrF_4 concentration less than 13.6 mol%. To study the local ionic structure of Na^+ ions in FLiNaK– ZrF_4 salts, the ^{23}Na NMR was performed on the salt samples exhibiting higher ZrF_4 concentration (Fig. 3).

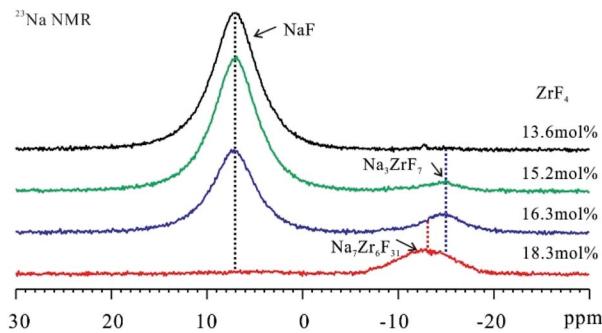


Fig. 3 ^{23}Na NMR spectra of FLiNaK– ZrF_4 ($13.6 \leq X_{\text{ZrF}_4} \leq 18.3$ mol%) eutectic solidification salts at ambient temperature. MAS spin rate in the mentioned experiments was set to 15 kHz.

The signal of NaF at 7.1 ppm tended to decrease and finally disappeared in the spectrum of ^{23}Na NMR with the rise of the ZrF_4 content in FLiNaK salt. Moreover, the disappearance of the NaF crystal phase was verified according to ^{19}F NMR results (Fig. S6†). It was therefore revealed that the complexation was developed between NaF and ZrF_4 phases when ZrF_4 component increased in FLiNaK. In addition, one broad signal at -15.0 ppm presented when 15.2 mol% of ZrF_4 component was introduced to FLiNaK salt. With the composition of ZrF_4 increasing further to 16.3 mol%, this signal remained and tended to increase. As suggested from this result, the Na-based coordinated crystal phases were formed in the mentioned salts. With the increase of the ZrF_4 concentration (18.3 mol%), this signal would shift to the low field (-12.9 ppm) and together with the half-peak width increased. This may be caused by the distribution of different new formed Na-based coordinated crystal phases.

To confirm the ion structure of the mentioned two types of Na-based complexes, NaF– ZrF_4 (40.5 mol%) and NaF– ZrF_4 (20 mol%) binary systems were prepared for comparison. Fig. 4 illustrates the ^{23}Na MAS NMR spectra for FLiNaK– ZrF_4 (16.3 mol%) and FLiNaK– ZrF_4 (18.3 mol%) salts. Obviously, one

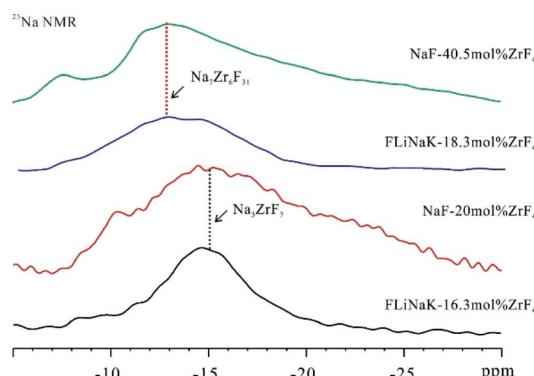


Fig. 4 ^{23}Na solid-state MAS NMR spectra of FLiNaK– ZrF_4 (16.3 mol%) and FLiNaK– ZrF_4 (18.3 mol%) eutectic solidification salt and binary salts like NaF– ZrF_4 (20 mol%) and NaF– ZrF_4 (40.5 mol%) at ambient temperature. MAS spin rate in the mentioned experiments was set to 15 kHz.



broad peak at -15.0 ppm was present in FLiNaK-ZrF_4 (16.3 mol%) salt.

As inspired by Thoma *et al.*,²¹ Na_3ZrF_7 crystals should exist in NaF-ZrF_4 (20 mol%), so the ^{23}Na NMR signal at -15.0 ppm belonged to Na_3ZrF_7 crystal phase. Moreover, a broad signal at nearly -12.9 ppm was identified in NaF-ZrF_4 (40.5 mol%) salt, which was reported to have $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystal phase.²¹ Thus, the low field shift of the broad signal might be attributed to the formation of $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystal phase in FLiNaK-ZrF_4 (18.3 mol%) salt.

3.3 Phase transition of K_3ZrF_7 crystals

With the formation of Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystals, a phase transition of K_3ZrF_7 crystals was also observed with the increase in the ZrF_4 concentration. Fig. 5 presents the ^{19}F NMR spectra of FLiNaK-ZrF_4 ($13.6 \leq X_{\text{ZrF}_4} \leq 18.3$ mol%) eutectic salts. Three signals with different full widths at half maximum (FWHM), *i.e.*, F'_1 , F''_1 and F'''_1 , were located at -35.0 ppm, -36.9 ppm and -37.8 ppm in FLiNaK-ZrF_4 (13.6 mol%) system, respectively. With the increase in the concentration of ZrF_4 , the strength of F'_1 site decreased F'_1 rapidly, and that of F''_1 and F'''_1 sites increased. Moreover, the new F''_1 and F'''_1 signals were observed to be broadened compared with F'_1 . In accordance with the existing reports,^{22–24} three crystal phases could exist in K_3ZrF_7 , *i.e.*, orthorhombic, tetragonal and disordered cubic lattice. Formation of the disordered cubic crystal phase always leads to the reduction of the crystallinity, and the broadening of the NMR signals should originate from the formation of the disordered crystal phase in the eutectic salt. Thus, for the higher ZrF_4 systems, *i.e.*, 16.3 mol% and 18.3 mol% of ZrF_4 within FLiNaK , K_3ZrF_7 was a disordered cubic crystal. According to the gradual increase of F''_1 and F'''_1 signals and the gradual decrease of F'_1 signals with the increase of ZrF_4 contents, it can be deduced that the K_3ZrF_7 phase transition was changed from orthorhombic lattice into disordered cubic lattice.

The formation of different $[\text{Zr}_x\text{F}_y]^{4x-y}$ ionic structures and crystal phases during the solidification of the salt melts could be discussed. After heating into melts, Zr^{4+} ions tended to form a coordinated structure with F^- ions as $[\text{ZrF}_7]^{3-}$ or $[\text{ZrF}_6]^{2-}$, which is similar to the results that $[\text{AF}_7]^{3-}$, $[\text{AF}_8]^{4-}$ and $[\text{AF}_9]^{5-}$

coexisted in LiF-AF_4 ($\text{A} = \text{Th}^{4+}$ or U^{4+}) molten salt.^{25,26} When cooling down, K_3ZrF_7 crystal phase would be formed first as impacted by the weak bonding energy of K^+ and F^- ions. The absence of $[\text{ZrF}_6]^{2-}$ could be attributed to the instability of such Zr-F coordination in solid phase. With the concentration of ZrF_4 increasing continuously, Na^+ ions would be involved in the formation of the new crystal phases and cause Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$ crystal phases to appear. This result was well consistent with the existing results that diffusion coefficients of K^+ , Na^+ and Li^+ successively increased within molten AF-ZrF_4 systems ($\text{A}^+ = \text{Li}^+$, Na^+ , K^+).²⁷ Moreover, the disordered cubic crystal phase of K_3ZrF_7 was formed as the ZrF_4 contents increased in FLiNaK . The formation of such disordered structure should account for the broadening of the ^{19}F NMR signals of the characteristic signals of K_3ZrF_7 in the FLiNaK-ZrF_4 eutectic salts exhibiting higher ZrF_4 concentration.

4 Conclusions

In brief, the evolution of the ionic structure of $[\text{Zr}_x\text{F}_y]^{4x-y}$ with the increase in ZrF_4 compositions in FLiNaK solidification salts was investigated specifically with NMR method. As indicated from the results, the complexation reactions of ZrF_4 tended to proceed as $\text{KF} > \text{NaF} > \text{LiF}$. Moreover, the predominant species (*e.g.*, K_3ZrF_7 , Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$) were formed, and the distribution of the mentioned species varied with the amount of ZrF_4 compositions from 0 to 18.3 mol%. To begin with, the content of K_3ZrF_7 was increased to maximum when ZrF_4 content was 13.6 mol%, then gradually decreased. However, the contents of Na_3ZrF_7 and $\text{Na}_7\text{Zr}_6\text{F}_{31}$ was increased when ZrF_4 contents was more than 13.6 mol%. Moreover, the phase of K_3ZrF_7 was observed to transit from orthorhombic lattice into disordered cubic lattice in the samples exhibiting higher ZrF_4 concentration, thereby causing the well-ordered crystal lattices to be destructed. The mentioned results present a further insight into the local structure of ZrF_4 -based molten salt and glasses systems.

Author contributions

Rongshan Lan: investigation; formal analysis; writing – original draft; Yiyang Liu: methodology; Ling Han: ICP-OES measurement analysis; Jing Yang and Huiqin Yin: resources of original materials; Min Ge and Yuan Qian: supervision; Hongtao Liu and Xiaobin Fu: conceptualization; writing – review and editing; funding acquisition.

Conflicts of interest

There are no conflicts to declare.

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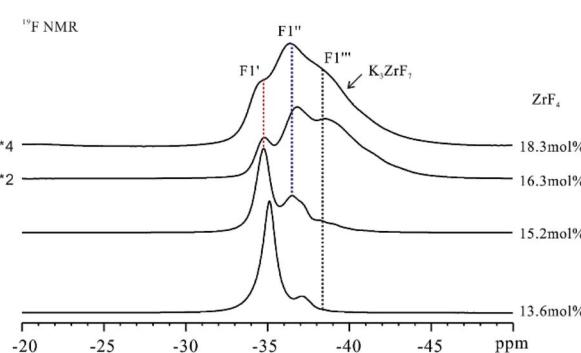


Fig. 5 ^{19}F NMR spectra of FLiNaK-ZrF_4 ($13.6 \leq X_{\text{ZrF}_4} \leq 18.3$ mol%) eutectic solidification salts at ambient temperature. MAS spin rate in the mentioned experiments was set to 15 kHz.



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References

- 1 C. W. Forsberg, P. F. Peterson and P. S. Pickard, *Nucl. Technol.*, 2003, **144**, 289–302.
- 2 J. Serp, M. Allibert, O. Beneš, S. Delpech, O. Feynberg, V. Ghetta, D. Heuer, D. Holcomb, V. Ignatiev, J. L. Kloosterman, L. Luzzi, E. Merle-Lucotte, J. Uhlir, R. Yoshioka and Z. M. Dai, *Prog. Nucl. Energy*, 2014, **77**, 308–319.
- 3 J. C. Gehin and J. J. Powers, *Nucl. Technol.*, 2016, **194**, 152–161.
- 4 H. Xu, Z. Dai and X. Cai, *Nucl. Phys. News*, 2014, **24**, 24–30.
- 5 M. Shen, H. Peng, M. Ge, C. Y. Wang, Y. Zuo and L. D. Xie, *RSC Adv.*, 2015, **5**, 40708–40713.
- 6 H. Peng, M. Shen, Y. Zuo, H. Y. Fu and L. D. Xie, *J. Nucl. Mater.*, 2018, **5**, 256–264.
- 7 H. Peng, Y. L. Song, N. Ji, L. D. Xie, W. Huang and Y. Gong, *RSC Adv.*, 2021, **11**, 18708–18716.
- 8 W. R. Grimes. *Reactor chemistry division annual proddress report*, Washington, USA, 7th edn, 1962.
- 9 L. M. Toth, A. S. Quist and G. E. Boyd, *J. Phys. Chem.*, 1973, **77**, 1384–1388.
- 10 V. Dracopoulos, J. Vagelatos and G. N. Papatheodorou, *J. Chem. Soc., Dalton Trans.*, 2001, **7**, 1117–1122.
- 11 A. L. Rollet and C. Bessada, *Annu. Rep. NMR Spectrosc.*, 2013, **78**, 149–207.
- 12 A. L. Rollet, H. Matsuura and C. Bessada, *Dalton Trans.*, 2014, **44**, 522–529.
- 13 X. M. Dou, D. Mohan, C. U. Pittman and S. Yang, *Chem. Eng. J.*, 2012, **198**, 236–245.
- 14 A. S. Vorob'ev, A. V. Suzdaltsev, P. S. Pershin, A. E. Galashev and Y. P. Zaikov, *J. Mol. Liq.*, 2020, **299**, 112241.
- 15 O. Pauvert, D. Zanghi, M. Salanne, C. Simon, A. Rakhmatullin, H. Matsuura, Y. Okamoto, F. Vivet and C. Bessada, *J. Phys. Chem. B*, 2010, **114**, 6472–6479.
- 16 J. F. Stebbins, *Modern Methods in Solid-state NMR*, ed. P. Hodgkinson, RSC, Cambridge, UK, 2018, ch. 9, pp. 262–288.
- 17 R. E. Youngman and S. Sabyasachi, *Solid State Nucl. Magn. Reson.*, 2005, **27**, 77–89.
- 18 C. Legein, F. Fayon, C. Martineau, M. Body, J.-Y. Buzare', D. Massiot, E. Durand, A. Tressaud, A. Demourgues, O. Pe'ron and B. Boulard, *Inorg. Chem.*, 2006, **45**, 10636–10641.
- 19 Y. Y. Liu, R. S. Lan, C. W. Dong, K. Wang, X. B. Fu, H. T Liu, Y. Qian and J. Q. Wang, *J. Phys. Chem. C*, 2021, **125**(8), 4704–4709.
- 20 A. Rakhmatullin, M. Boča, J. Mlynáriková, E. Hadzimová, Z. Vasková, I. B. Polovov and M. Mičušík, *J. Fluorine Chem.*, 2018, **208**, 24–35.
- 21 C. J. Barton, W. R. Grimes, H. Insley, R. E. Moore and R. E. Thoma, *J. Phys. Chem.*, 1958, **62**, 665–676.
- 22 E. C. Reynhardt, J. C. Pratt, A. Watton and H. E. Petch, *J. Phys. C: Solid State Phys.*, 1981, **14**, 4701–4715.
- 23 M. T. Dov, M. C. Caracoche, A. M. Rodríguez, J. A. Martínez, P. C. Rivas and A. R. L. García, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1989, **40**, 11258–11263.
- 24 M. V. Gorev, M. S. Molokeev, A. V. Kartashev, E. I. Pogoreltsev, S. V. Mel'nikova, N. M. Laptash and I. N. Flerov, *J. Fluorine Chem.*, 2021, **241**, 10967.
- 25 X. J. Guo, H. L. Qian, J. X. Dai, W. H. Liu, J. T. Hu, R. F. Shen and J. Q. Wang, *J. Mol. Liq.*, 2019, **277**, 409–417.
- 26 C. Bessada, D. Zanghi, M. Salanne, A. Gil-Martin, M. Gibilaro, P. Chamelot, L. Massot, A. Nezu and H. Matsuura, *J. Mol. Liq.*, 2020, **307**, 112927.
- 27 M. Salanne, C. Simon, H. Groult, F. Lantelme, T. Goto and A. Barhoud, *J. Fluorine Chem.*, 2009, **130**, 61–66.

