

Cite this: *Chem. Sci.*, 2020, 11, 3986

All publication charges for this article have been paid for by the Royal Society of Chemistry

# Transforming colloidal Cs<sub>4</sub>PbBr<sub>6</sub> nanocrystals with poly(maleic anhydride-*alt*-1-octadecene) into stable CsPbBr<sub>3</sub> perovskite emitters through intermediate heterostructures†

Dmitry Baranov,<sup>a</sup> Gianvito Caputo,<sup>a</sup> Luca Goldoni,<sup>b</sup> Zhiya Dang,<sup>a</sup> Riccardo Scarfiello,<sup>c</sup> Luca De Trizio,<sup>a</sup> Alberto Portone,<sup>d</sup> Filippo Fabbri,<sup>d</sup> Andrea Camposeo,<sup>d</sup> Dario Pisignano<sup>b,de</sup> and Liberato Manna<sup>a</sup>

The preparation of strongly emissive CsPbBr<sub>3</sub> perovskite nanocrystals with robust surface passivation is a challenge in the field of lead halide perovskite nanomaterials. We report an approach to prepare polymer-capped CsPbBr<sub>3</sub> perovskite nanocrystals by reacting oleylammonium/oleate-capped Cs<sub>4</sub>PbBr<sub>6</sub> nanocrystals with poly(maleic anhydride-*alt*-1-octadecene) (PMAO). PMAO contains succinic anhydride units that are reactive towards the oleylamine species present on the Cs<sub>4</sub>PbBr<sub>6</sub> nanocrystals' surface and produces polysuccinamic acid, which, in turn, triggers the Cs<sub>4</sub>PbBr<sub>6</sub> to CsPbBr<sub>3</sub> conversion. The transformation occurs through the formation of Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures as intermediates, which are captured because of the mild reactivity of PMAO and are investigated by high-resolution electron microscopy. The Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures demonstrate a dual emission at cryogenic temperature with an indication of the energy transfer from Cs<sub>4</sub>PbBr<sub>6</sub> to CsPbBr<sub>3</sub>. The fully-transformed CsPbBr<sub>3</sub> NCs have high photoluminescence quantum yield and enhanced colloidal stability, which we attribute to the adhesion of polysuccinamic acid to the NC surface through its multiple functional groups in place of oleate and alkylammonium ligands. The PMAO-induced transformation of Cs<sub>4</sub>PbBr<sub>6</sub> NCs opens up a strategy for the chemical modification of metal halide NCs initially passivated with nucleophilic amines.

Received 7th February 2020

Accepted 20th March 2020

DOI: 10.1039/d0sc00738b

rsc.li/chemical-science

## Introduction

Nanocrystals (NCs) of cesium lead halides have recently emerged as a class of semiconductor materials promising for light-emitting applications.<sup>1–3</sup> The chemical reactivity of these NCs and the interconversion between the NCs of the two most studied bromides in this class, Cs<sub>4</sub>PbBr<sub>6</sub> and CsPbBr<sub>3</sub> perovskite, have been of interest since these NCs were first synthesized in the colloidal form.<sup>4–8</sup> The Cs<sub>4</sub>PbBr<sub>6</sub> → CsPbBr<sub>3</sub> conversion, which can be triggered using various reagents (for example, Prussian blue,<sup>9</sup> oleic acid,<sup>10</sup> PbBr<sub>2</sub>,<sup>7,11</sup> and water)<sup>12–14</sup> is an interesting approach to prepare emissive CsPbBr<sub>3</sub> NCs. For

example, Yin's group exploited heterogeneous water-mediated CsBr extraction from Cs<sub>4</sub>PbBr<sub>6</sub> NCs in hexane as a method for making luminescent CsPbBr<sub>3</sub>/SiO<sub>2</sub> or CsPbBr<sub>3</sub>/Ta<sub>2</sub>O<sub>5</sub> Janus-type heterostructures,<sup>13</sup> and branched CsPbBr<sub>3</sub> dodecapods.<sup>15</sup> Despite several reports on Cs<sub>4</sub>PbBr<sub>6</sub> → CsPbBr<sub>3</sub> transformation at the nanoscale, the nanocrystal intermediates of this reaction and the surface passivation and stability of the resulting CsPbBr<sub>3</sub> NCs have not been investigated.

Designing the Cs<sub>4</sub>PbBr<sub>6</sub> → CsPbBr<sub>3</sub> NC transformation in such a way that it delivers encapsulated CsPbBr<sub>3</sub> NCs with an enhanced stability is a promising approach for exploiting the Cs<sub>4</sub>PbBr<sub>6</sub> NC reactivity, as shown by the above mentioned studies of Yin's group.<sup>13</sup> The use of an organic polymer instead

<sup>a</sup>Nanochemistry Department, Istituto Italiano di Tecnologia, Via Morego 30, 16163 Genova, Italy. E-mail: dmitry.baranov@iit.it; liberato.manna@iit.it

<sup>b</sup>Analytical Chemistry Lab, Istituto Italiano di Tecnologia, Via Morego 30, 16163 Genova, Italy

<sup>c</sup>CNR NANOTEC, Institute of Nanotechnology, c/o Campus Ecotecnico, via Monteroni, 73100 Lecce, Italy

<sup>d</sup>NEST, Istituto Nanoscienze-CNR, Piazza S. Silvestro 12, I-56127 Pisa, Italy

<sup>e</sup>Dipartimento di Fisica "Enrico Fermi", Università di Pisa, Largo Bruno Pontecorvo 3, I-56127 Pisa, Italy

† Electronic supplementary information (ESI) available: Experimental details and procedures, EDS-STEM data, <sup>1</sup>H and <sup>1</sup>H-<sup>13</sup>C HSQC NMR spectra and discussion, annotated XRD patterns, PLQY spectra, FTIR and NIR absorbance spectra, tests of PMAO reactivity with powders of bulk Cs<sub>4</sub>PbBr<sub>6</sub> and amine-free Cs<sub>4</sub>PbBr<sub>6</sub> NCs, stability tests of CsPbBr<sub>3</sub>/PMAO NCs, HRTEM images of Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures, low-resolution TEM size analysis, PL maps and spectra of Cs<sub>4</sub>PbBr<sub>6</sub> NCs at 27 K, and time-resolved PL, micro-PL, and Raman spectra for the NC-PMAO blend (PDF). A video showing transformation of non-luminescent Cs<sub>4</sub>PbBr<sub>6</sub> NCs into green-emitting CsPbBr<sub>3</sub> NCs after addition of PMAO (MP4). See DOI: 10.1039/d0sc00738b.



of an inorganic oxide (e.g.  $\text{SiO}_2$  or  $\text{Ta}_2\text{O}_5$ ) shell would yield a NC-polymer blend which can be drop-cast, spin-coated or electro-spun, widening the range of available applications.<sup>16</sup> More generally, polymer encapsulation of  $\text{CsPbX}_3$  perovskite NCs ( $X = \text{Cl}, \text{Br}, \text{I}$ , and their mixtures) is promising because it has been shown to enhance the shelf-time of NCs by providing enhanced stability against moisture and photodegradation.<sup>17</sup> Interestingly, stability enhancement has been reported irrespective of whether polymer chains preserve the native  $\text{CsPbX}_3$  NC surface ligands as in the case of polystyrene<sup>17–19</sup> and poly(styrene-ethylene-butylene-styrene),<sup>17</sup> or whether the polymer adheres to the surface of  $\text{CsPbX}_3$  NCs as in the case of ammonium bromide-terminated polystyrene<sup>20</sup> or poly(*n*-butyl methacrylate) modified with zwitterionic sulfobetaine or phosphorylcholine functional groups.<sup>21</sup> Arguably, an ideal NC transformation of  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  in this context could be caused by a polymer which acts both as a reactant and a macromolecular surfactant,<sup>20</sup> minimizing the number of reagents and preparatory steps involved in the process.

In this work, we demonstrate that poly(maleic anhydride-1-*alt*-octadecene) (PMAO) can simultaneously trigger the  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  NC transformation and provide enhanced surface passivation to the resulting  $\text{CsPbBr}_3$  NCs. PMAO is a widely available co-polymer of 1-octadecene and maleic anhydride and has been extensively used for the surface functionalization of NCs.<sup>22–24</sup> In our experiments, upon mixing PMAO with oleylammonium/oleate-capped  $\text{Cs}_4\text{PbBr}_6$  NCs, the cyclic anhydride groups of PMAO react with oleylamine species, forming polysuccinamic acid (Fig. 1). Polysuccinamic acid destabilizes the NC surface by displacing both the amine and the oleate ligands and acidifies the reaction environment, thus triggering the formation of  $\text{CsPbBr}_3$  NCs (Fig. 1). The core chemistry of the NC transformation is summarized by the following chemical equation:  $\text{Cs}_4\text{PbBr}_6 + n\text{RNH}_2 + (-\text{R}'(\text{CHCO})_2\text{O}-)_n \rightarrow \text{CsPbBr}_3 + (-\text{R}'(\text{CHCOOH})(\text{CHCONHR})-)_n + 3\text{Cs}^+_{(\text{solvated})} + 3\text{Br}^-_{(\text{solvated})}$ , where  $\text{R} = \text{oleyl}$ ,  $\text{R}' = \text{octadecenyl}$ ,

and the ratio between oleylamine molecules and anhydride units is assumed to be 1 : 1 for simplicity. The extent of the transformation is tunable by varying the amount of added PMAO, enabling the investigation of the transformation intermediates, which consist of  $\text{Cs}_4\text{PbBr}_6$ - $\text{CsPbBr}_3$  heterostructures. The fully-transformed  $\text{CsPbBr}_3$  NCs are bright emitters and retain their green emission for four weeks of storage under ambient conditions in air, even after one washing cycle with ethyl acetate (a solvent which typically causes the degradation of oleylammonium/oleate-capped  $\text{CsPbBr}_3$  NCs within hours or days). The increase in the stability of  $\text{CsPbBr}_3$  NCs synthesized from  $\text{Cs}_4\text{PbBr}_6$  and PMAO is attributed to the adhesion of polysuccinamic acid to the NC surface through its multiple functional groups.

## Results and discussion

### $\text{Cs}_4\text{PbBr}_6$ NCs and their transformation with PMAO in solution

The synthesis of the initial  $\text{Cs}_4\text{PbBr}_6$  NCs was performed in air, *via* the hot injection of cesium oleate into the solution of lead(II) bromide dissolved in a mixture of oleylamine and oleic acid in 1-octadecene,<sup>7</sup> as detailed in Section S1 of the ESI.† The synthesis is similar to that of  $\text{CsPbBr}_3$  NCs,<sup>5</sup> except that it is performed at a higher concentration of oleylamine and oleic acid with respect to lead ( $[\text{oleylamine}] : [\text{oleic acid}] : [\text{PbBr}_2] \sim 0.63 : 0.31 : 0.027 \text{ M}$ ). Such reaction conditions favor the formation of a Pb-poor  $\text{Cs}_4\text{PbBr}_6$  phase over the  $\text{CsPbBr}_3$  phase, as detailed previously.<sup>25</sup> The synthesis delivers batches of Cs-rich rhombohedral  $\text{Cs}_4\text{PbBr}_6$  NCs with a narrow size distribution and an average diameter in the range from 10 to 16 nm (Fig. 2a, b, and S1–S5†).  $^1\text{H}$  and  $^1\text{H}$ - $^{13}\text{C}$  heteronuclear single quantum coherence (HSQC) nuclear magnetic resonance (NMR) experiments established that  $\text{Cs}_4\text{PbBr}_6$  NCs are passivated by a mixture of oleylammonium oleate and neutral oleylamine with a ligand ratio of  $\sim 3 : 2$  between (oleylamine + oleylammonium) : oleate species (see Section S3 and Fig. S6–S9 of the ESI for details†).

These  $\text{Cs}_4\text{PbBr}_6$  NCs can react with PMAO in toluene, fully or partially transforming into  $\text{CsPbBr}_3$  NCs, depending on the amount of the added polymer (Fig. 2c). The reaction of  $\text{Cs}_4\text{PbBr}_6$  NCs with PMAO typically starts within a few minutes after the addition of PMAO at room temperature (Movie S1 and Fig. S10†) and can be accelerated by mild heating of the reaction mixture (up to 80 °C). It is important to highlight here that heating up the NCs alone to 80 °C without the addition of PMAO does not trigger any transformation (Fig. S11†). The fully-transformed  $\text{CsPbBr}_3$  NCs have a narrow size distribution, as inferred from their self-organization into ordered close-packed monolayers on a carbon-coated TEM copper grid (Fig. 2a). The XRD patterns of the initial NCs, partially- and completely-transformed samples are shown in Fig. 2b. Following the transformation, the XRD reflections of the rhombohedral  $\text{Cs}_4\text{PbBr}_6$  crystal structure gradually disappeared, and peaks characteristic of the orthorhombic  $\text{CsPbBr}_3$  perovskite phase emerged (Fig. 2b, see Fig. S12† for peak assignment). The progression of the reaction was monitored by steady-state UV-Vis absorption spectroscopy



**Fig. 1** Schematic representation of the  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  NC transformation induced by PMAO. Oleylamine species from the NC surface react with the cyclic anhydride rings of PMAO, forming polysuccinamic acid. The removal of oleylamine-based and oleate ligands destabilizes the NC surface, and the formation of polysuccinamic acid increases the acidity of the medium, triggering the  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  transformation (see the text for the chemical equation). The resulting  $\text{CsPbBr}_3$  NCs are stabilized by the polysuccinamic acid in place of the original ligands.





Fig. 2 (a) Low-magnification TEM images of the initial  $\text{Cs}_4\text{PbBr}_6$  NCs (average diameter  $10 \pm 1.5$  nm) and fully-transformed  $\text{CsPbBr}_3$  NCs (average edge length  $8 \pm 0.4$  nm) after their reaction with PMAO in toluene. (b) XRD patterns of the initial  $\text{Cs}_4\text{PbBr}_6$  NCs and of the partially- and fully-transformed ones. Top and bottom stick patterns are those of the reference bulk compounds: rhombohedral  $\text{Cs}_4\text{PbBr}_6$  (pattern ID 04-015-9683, ICSD code 162158)<sup>27</sup> and orthorhombic  $\text{CsPbBr}_3$  (pattern ID 96-451-0746, COD code 4510745).<sup>28</sup> The broad peak at  $\sim 20^\circ$  is due to PMAO. (c) Optical absorption (left panel) and emission (right panel) spectra of toluene solutions of initial  $\text{Cs}_4\text{PbBr}_6$  NCs (blue curve), PMAO (orange curve), and partially- (black curve), and fully-transformed (green curve) NCs. The inset in the left panel shows photographs of the samples under excitation with a 365 nm lamp demonstrating visible green PL of partially- and fully-transformed NC samples.

(Fig. 2c, left panel, Fig. S10<sup>†</sup>) in which the disappearance of the  $\sim 314$  nm peak characteristic of  $\text{Cs}_4\text{PbBr}_6$ <sup>7</sup> and the appearance of the  $\sim 510$  nm band edge absorption of  $\text{CsPbBr}_3$  are evident. The transformation was also tracked by steady-state photoluminescence (PL) spectroscopy, through the appearance of a cyan emission ( $\lambda_{\text{max}} \sim 475\text{--}480$  nm) in the early stages of the reaction (Fig. S10<sup>†</sup>). The absolute PL quantum yield (PLQY) of the samples transformed in air was measured to be  $\sim 19\%$  (partially transformed), and  $\sim 25\%$  (fully transformed). On the other hand, when the transformation was performed under an inert atmosphere, the sample had a 69% PLQY (Fig. S13–S15<sup>†</sup>). Such a value is comparable to those reported for other  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  chemical transformations of NCs: 47% (ref. 7) and 62% (ref. 11) *via* the addition of solid  $\text{PbBr}_2$  at elevated temperatures, and 75% (ref. 12) upon reaction with  $\text{H}_2\text{O}$ . The lower PLQY of the samples transformed in air is attributed to the presence of electron traps formed as a result of sample exposure to atmospheric  $\text{O}_2$ . Similar results have been reported by Rodà *et al.* who observed PL dimming in oxygen-exposed  $\text{CsPbBr}_3$  nanocubes.<sup>26</sup>

#### Rationalization of the observed reactivity between PMAO and $\text{Cs}_4\text{PbBr}_6$ NCs

PMAO is a copolymer of octadecene-1 and maleic anhydride, and it consists of repeating units composed of a saturated

hydrocarbon chain and a cyclic succinic anhydride ring (Fig. 1). PMAO has a negligible reactivity towards inorganic salts such as  $\text{Cs}_4\text{PbBr}_6$ , as confirmed in a control experiment on finely ground powder of bulk  $\text{Cs}_4\text{PbBr}_6$  (Fig. S16 and S17<sup>†</sup>). However, the succinic anhydride rings of PMAO feature acyl groups that are reactive towards nucleophilic reagents such as water and primary amines (yielding, in the latter case, either succinamic acid at room temperature<sup>23,29–31</sup> or cyclic imides at high temperatures<sup>31–33</sup>). The presence of a significant amount of water as a potential reactant towards PMAO in the  $\text{Cs}_4\text{PbBr}_6$  NC samples was ruled out based on FTIR and NIR characterization (Fig. S18 and S19<sup>†</sup>). On the other hand, the ligand shell of  $\text{Cs}_4\text{PbBr}_6$  NCs contains partially-protonated oleylamine (Section S3 and Fig. S6–S9<sup>†</sup>). In analogy with the widely studied oleylammonium/oleate-capped  $\text{CsPbBr}_3$  NCs,<sup>25,34–36</sup> the ligands on the surface of  $\text{Cs}_4\text{PbBr}_6$  NCs are likely to exist in a dynamic equilibrium between neutral and protonated species (oleylamine and oleylammonium, respectively). Thus neutral oleylamine is always available in the NC solution. Neutral oleylamine is a nucleophile with a documented reactivity towards linear and cyclic anhydrides,<sup>37,38</sup> and polymaleic anhydride derivatives.<sup>39,40</sup> The reaction between neutral oleylamine and PMAO in the absence of NCs causes broadening of the vinyl hydrogen resonance of the oleyl chain in the  $^1\text{H}$  NMR spectrum due to the attachment of small oleylamine molecules to PMAO



macromolecules (the specified  $M_w$  of PMAO is  $\sim 30\,000$ – $50\,000$  g mol $^{-1}$ , which roughly corresponds to  $\sim 80$ – $150$  succinic anhydride-octadecene subunits) (Fig. S20–S22†). The addition of neutral oleylamine to cyclic anhydride produces a succinamic acid derivative, as was confirmed by  $^1\text{H}$  and  $^1\text{H}$ - $^{13}\text{C}$  HSQC NMR in a control reaction (Fig. S23†). Therefore, the formation of polysuccinamic acid (Fig. 1) is expected upon mixing of PMAO with oleylammonium/oleate capped  $\text{Cs}_4\text{PbBr}_6$  NCs. The key role of oleylamine species in the  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  NC transformation was further verified by a control reaction between PMAO and oleylamine-free  $\text{Cs}_4\text{PbBr}_6$  NCs (synthesized with tri-*n*-octylphosphine oxide (TOPO) and oleic acid<sup>41</sup>). The  $\text{Cs}_4\text{PbBr}_6$  NCs synthesized with TOPO and oleic acid were found to be unreactive towards PMAO (Fig. S27 and S28†).

In summary, the removal of oleylamine from the surface of  $\text{Cs}_4\text{PbBr}_6$  NCs destabilizes them, while polysuccinamic acid acidifies the reaction environment. Surface destabilization and acidic environments are both general conditions that are known to cause the  $\text{Cs}_4\text{PbBr}_6 \rightarrow \text{CsPbBr}_3$  transformation.<sup>10,25,42</sup> The stoichiometry of the transformation is balanced by a nominal removal of 3 equivalents of CsBr from 1 equivalent of  $\text{Cs}_4\text{PbBr}_6$ , yet we have not experimentally detected crystalline CsBr by XRD or high-resolution TEM (HRTEM). This discrepancy is tentatively rationalized by solvation of  $\text{Cs}^+$  and  $\text{Br}^-$  ions by oleate and polysuccinamic acid species, similar to the previously reported dissociation of CsBr in dimethylformamide in the presence of the polyacrylic acid co-polymer.<sup>43</sup> Eventually, our  $^1\text{H}$  NMR analysis also revealed that the final NCs were capped solely by polysuccinamic acid, indicating the displacement of both the oleate and amine/ammonium ligands from the NC surface

upon transformation (see the discussion in Section S9 and Fig. S20–S26†).

### Enhanced stability of the $\text{CsPbBr}_3/\text{PMAO}$ NCs

The fully-transformed  $\text{CsPbBr}_3$  NCs formed an optically clear solution in toluene. These  $\text{CsPbBr}_3$  NCs possessed an enhanced stability compared to the polymer-free, ligand-capped  $\text{CsPbBr}_3$  NCs directly synthesized by following the cesium oleate/lead(II) bromide route.<sup>5,25</sup> Such enhanced stability was demonstrated by the fact that the NCs retained their green emission after four weeks of storage under ambient conditions in air (Fig. S29†), even after undergoing a washing cycle of precipitation/redispersion with ethyl acetate (Fig. S30 and S31†), while the polymer-free  $\text{CsPbBr}_3$  NCs aggregated within hours or days after undergoing a similar washing procedure. Another indicator of the increased stability is the observation that the  $\text{CsPbBr}_3/\text{PMAO}$  NCs could be concentrated or diluted over  $\sim 5$  orders of magnitude range of concentrations, from  $\sim 26$  mg ml $^{-1}$  to  $\sim 1 \times 10^{-4}$  mg ml $^{-1}$ , without any loss of optical transparency or PL emission (Fig. S32†). The increase in the stability of the fully-transformed  $\text{CsPbBr}_3/\text{PMAO}$  NCs is in agreement with prior reports on  $\text{CsPbBr}_3$  NCs blended with PMAO<sup>44,45</sup> or with the related dodecyl-grafted-poly(isobutylene-*alt*-maleic-anhydride).<sup>46</sup> Our hypothesis is that the binding of polysuccinamic acid through its multiple functional groups to the NC surface, in place of the standard ligands used in the direct synthesis of  $\text{CsPbBr}_3$  NCs (as discussed above and in Section S9, Fig. S20–S26†), is the origin of this enhancement. To test this hypothesis, we compared the solvodynamic diameters of PMAO and



**Fig. 3** (a–e) HRTEM images of  $\text{Cs}_4\text{PbBr}_6$ – $\text{CsPbBr}_3$  heterostructures formed upon partial conversion of  $\text{Cs}_4\text{PbBr}_6$  NCs with PMAO (scale bars are 2 nm).  $\text{Cs}_4\text{PbBr}_6$  domains are shaded in blue, and  $\text{CsPbBr}_3$  domains are shaded in green; (f) a magnified view of (c) and (g) the corresponding FFT image; and (h) ball-and-stick atomic model of the epitaxial interface built using VESTA software (ver. 3.4.6, the atoms are depicted as spheres with radii corresponding to 40% of actual atomic radii).<sup>47</sup> Cs atoms in the model are colored in two different colors for clarity: in cyan for  $\text{Cs}_4\text{PbBr}_6$  and green for  $\text{CsPbBr}_3$ .



CsPbBr<sub>3</sub>/PMAO NCs (washed once with ethyl acetate) determined by dynamic light scattering ( $\sim 1.7 \pm 1.2$  nm and  $\sim 11.2 \pm 0.9$  nm, respectively, Fig. S33 and S34†) with the sizes of the inorganic CsPbBr<sub>3</sub> cores from the TEM analysis of the same sample ( $\sim 7$  nm edge length, Fig. S35†). The larger solvodynamic diameter of CsPbBr<sub>3</sub>/PMAO NCs in solution compared to the CsPbBr<sub>3</sub> NC edge length from TEM is explained by the PMAO wrapping and NC tumbling in solution (the diagonal of a cube with a 7 nm edge length is  $\sim 12$  nm). The lack of a substantial increase in the solvodynamic diameter of CsPbBr<sub>3</sub>/PMAO NCs is interpreted as an indicator of PMAO wrapping around NCs, supporting the hypothesis about the origin of increased NC stability. In addition, the relatively small solvodynamic diameter of CsPbBr<sub>3</sub>/PMAO NCs indicates that PMAO molecules do not bind multiple NCs together.

### Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures

The Cs<sub>4</sub>PbBr<sub>6</sub> → CsPbBr<sub>3</sub> NC transformation with PMAO is relatively slow at room temperature. This enabled the observation of NC intermediates consisting of Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures (Fig. 3), which were investigated by HRTEM (Fig. 3a-e). In one of the partially-transformed samples we observed NCs with different degrees of conversion (Fig. 3a-e). The heterostructures displayed a variety of interfaces between Cs<sub>4</sub>PbBr<sub>6</sub> and CsPbBr<sub>3</sub>, some adopting an epitaxial relationship, some not (analysis of the cases is shown in Fig. S36†). For example, the heterostructure shown in Fig. 3c, analyzed in detail in Fig. 3f-h, is characterized by an epitaxial relationship adopted by the two domains, as indicated by the overlap of the spots from the planes of the two crystal structures in fast Fourier transform (FFT, Fig. 3g) of the real space image. The  $<5\%$  mismatch between the atomic spacing of the two domains [ $d(3\bar{1}2)_{\text{CsPbBr}_3} = 2.37$  Å,  $d(6\bar{3}0)_{\text{Cs}_4\text{PbBr}_6} = 2.29$  Å] leads to a slight bending of the planes, as labeled by the dashed lines in Fig. 3f. This bending also indicates that the atomic planes of Cs<sub>4</sub>PbBr<sub>6</sub> domains on the two sides of the CsPbBr<sub>3</sub> domain are rotated by a small angle. The rotation gives rise to extended diffraction spots in the FFT image, instead of single sharp spots that would otherwise appear for a single crystal. Considering an orthorhombic phase for CsPbBr<sub>3</sub> (ICSD: 97851,  $a = 8.207$  Å,  $b = 8.255$  Å,  $c = 11.759$  Å), the epitaxial relationship between the two domains can be described as follows: CsPbBr<sub>3</sub> [02 $\bar{1}$ ] || Cs<sub>4</sub>PbBr<sub>6</sub> [001], and CsPbBr<sub>3</sub> (112) || Cs<sub>4</sub>PbBr<sub>6</sub> (030) (see Fig. 3h).

The low-magnification TEM images of the two NC samples were analyzed to quantify changes in the NC dimensions before and after the transformation (Fig. S37-S42†). For example, a sample of  $10.1 \text{ nm} \pm 1.4$  nm diameter Cs<sub>4</sub>PbBr<sub>6</sub> NCs transformed into  $8 \text{ nm} \pm 0.4$  nm edge length CsPbBr<sub>3</sub> NCs (Fig. 2a). The Scherrer analysis of the XRD patterns of the same sample before and after the transformation indicated a reduction in the crystallite size from  $16.1 \pm 1.8$  nm to  $12.5 \pm 2.6$  nm, in agreement with the TEM analysis (larger dimensions from XRD as compared to TEM are due to the differences between techniques and analyses). In another sample,  $15.7 \text{ nm} \pm 2.6$  nm Cs<sub>4</sub>PbBr<sub>6</sub> NCs transformed into  $12 \text{ nm} \pm 1.9$  nm NCs (dimensions from TEM). If one assumes that such transformation does

not proceed by dissolution-recrystallization, but simply by the gradual removal of CsBr from each individual spherical NC of Cs<sub>4</sub>PbBr<sub>6</sub>, converting it to a cube-shaped NC of CsPbBr<sub>3</sub>, then by volume contraction the resulting CsPbBr<sub>3</sub> NCs should have an edge length of 6 nm in TEM (9.5 nm in the second sample), which is  $\sim 2$  nm smaller than the obtained value (Table S1†). Hence, dissolution-recrystallization processes should also play an important role in this transformation. A similar mechanism has been previously invoked to rationalize the inverse NC transformation (from CsPbBr<sub>3</sub> to Cs<sub>4</sub>PbBr<sub>6</sub>).<sup>8,10</sup>

The PL of the partially-converted sample containing Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> heterostructures was surveyed at room and



Fig. 4 PL maps of (a) partially-transformed Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> NCs at room temperature, and the inset shows the PL excitation spectrum at  $\sim 504$  nm (indicated by a white dashed line in the PL map); (b) partially-transformed Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> NCs at  $\sim 35$  K, and the inset shows the low intensity region around 376 nm; (c) PL spectra of Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> NCs at 292 K (left panel) and 35 K (right panel) collected under  $\sim 312$  nm (red curve) and  $\sim 402$  nm (green curve) excitation.



cryogenic temperatures (see Section S14 of the ESI† for experimental details) because their optical properties are unknown to date. The results are presented in Fig. 4a and b as excitation-emission maps (PL maps). The room temperature ( $T \sim 292$  K) PL map of the partially-converted sample contains a single emission peak of CsPbBr<sub>3</sub> at  $\sim 504$  nm (Fig. 4a). The CsPbBr<sub>3</sub> emission has a broad PL excitation spectrum (inset in Fig. 4a) with a dip at  $\sim 314$  nm characteristic of Cs<sub>4</sub>PbBr<sub>6</sub> absorption. Upon cooling to  $T \sim 35$  K, the PL map shows two emission peaks (Fig. 4b): an intense peak at  $\sim 513$  nm and a weak peak at  $\sim 376$  nm (inset in Fig. 4b). The  $\sim 513$  nm peak is an emission feature of CsPbBr<sub>3</sub>, red-shifted from  $\sim 504$  nm as a result of cooling.<sup>48,49</sup> The  $\sim 376$  nm emission with narrow excitation at  $\sim 313$  nm is assigned to Cs<sub>4</sub>PbBr<sub>6</sub> because it matches with previously reported cryogenic PL spectra of bulk Cs<sub>4</sub>PbBr<sub>6</sub> (ref. 50) and Cs<sub>4</sub>PbBr<sub>6</sub> aggregates in CsBr.<sup>51</sup> This assignment was further confirmed by collecting the PL map of the as-synthesized Cs<sub>4</sub>PbBr<sub>6</sub> NCs at  $T \sim 27$  K (Fig. S43†). At 27 K, the emission of the as-synthesized Cs<sub>4</sub>PbBr<sub>6</sub> NCs is dominated by a peak at  $\sim 376$  nm surrounded by weaker features due to various electronic transitions in Pb<sup>2+</sup> ions.<sup>52–54</sup> The as-synthesized Cs<sub>4</sub>PbBr<sub>6</sub> NCs are not emissive at room temperature and, besides the discussed  $\sim 376$  nm emission, are non-emissive up to the detection limit of 1600 nm when cooled (Fig. S44†).

The dual emission of partially-transformed NCs provides an opportunity to probe the energy transfer between Cs<sub>4</sub>PbBr<sub>6</sub> and CsPbBr<sub>3</sub>. Fig. 4c shows a comparison between pairs of emission spectra for the partially-converted sample collected at two different temperatures (292 K and 35 K) and two different excitation energies: one matching with Cs<sub>4</sub>PbBr<sub>6</sub> absorption ( $\lambda_{\text{exc}} \sim 312$  nm) and one below it ( $\lambda_{\text{exc}} \sim 402$  nm, only CsPbBr<sub>3</sub> absorbs). At 292 K (Fig. 4c, left panel), only CsPbBr<sub>3</sub> emits, regardless of excitation energy, and its emission is quenched by a factor of  $\sim 2.3$  after changing the excitation energy from  $\sim 402$  nm to  $\sim 312$  nm. This quenching is attributed to the attenuation of  $\sim 312$  nm excitation due to absorption by Cs<sub>4</sub>PbBr<sub>6</sub> and an excitation-

dependent PL efficiency.<sup>55</sup> At 35 K, both materials emit, and the CsPbBr<sub>3</sub> emission is quenched by a smaller factor of  $\sim 1.3$  (Fig. 4c, right panel). We can assign the lower quenching of CsPbBr<sub>3</sub> emission at 35 K to the energy transfer from Cs<sub>4</sub>PbBr<sub>6</sub>, which indeed is favored due to the overlap between the emission of the donor (Cs<sub>4</sub>PbBr<sub>6</sub>) and the absorption of the acceptor (CsPbBr<sub>3</sub>). These initial observations make Cs<sub>4</sub>PbBr<sub>6</sub>-CsPbBr<sub>3</sub> NCs a promising platform for future spectroscopic studies of the energy flow between lead halide perovskites and related compounds.

### Reactivity of Cs<sub>4</sub>PbBr<sub>6</sub> NC samples with PMAO in drop-cast films

The reaction described above can also proceed inside a polymer film (as was confirmed by *in situ* Raman spectroscopy, see Fig. S45†), which makes its investigation relevant for the emerging application of blends between PMAO and oleylammonium/oleate-capped perovskite NCs in light-emitting diodes.<sup>44,45</sup> From this point of view, the Cs<sub>4</sub>PbBr<sub>6</sub> to CsPbBr<sub>3</sub> transformation is an indicator of amine-anhydride reactivity, and its kinetics can be studied *in situ* by steady-state and time-resolved PL. Fig. 5 shows the results of the *in situ* PL measurements from a macroscopic area ( $\sim 2$  mm excitation spot size) of the film made by quick drop-casting of a freshly prepared PMAO-Cs<sub>4</sub>PbBr<sub>6</sub> NCs blend. Green PL develops within the first few minutes in the drop-cast film, and reaches a stable intensity and position ( $\sim 510$  nm, full width at half maximum of 18 nm) after  $\sim 2$  hours (Fig. 5a and b), indicating the timescale of the complete conversion. Both the PL intensity and the absorbance of the film (at 405 nm, the wavelength of the CW laser used for excitation) increase over the course of the transformation, with a characteristic time constant of about  $\sim 10$  minutes (Fig. 5c). Similar kinetics were obtained by *in situ* micro-PL performed with a confocal fluorescence microscope (Fig. S46†), suggesting that the transformation proceeds uniformly across the blend. The PLQY in the film remains almost constant at  $\sim 20\%$



Fig. 5 Tracking the Cs<sub>4</sub>PbBr<sub>6</sub> → CsPbBr<sub>3</sub> NC transformation in a drop-cast film by PL spectroscopy. (a) PL spectrum of the fully transformed CsPbBr<sub>3</sub> NCs, peaking at  $\sim 510$  nm. (b) Spectrally-resolved temporal evolution of the PL spectrum on a minute scale, for  $\sim 120$  minutes. (c) Time evolution of the integrated PL intensity fitted with first-order kinetics (solid black line, time  $\sim 10$  min). The inset shows time-dependent absorbance at 405 nm over the course of the transformation. (d) Time-dependent PLQY of the drop-cast film. (e) PL intensity map showing the picosecond temporal behavior of the emission intensity of the drop-cast films. (f) Temporal evolution of the longer PL decay lifetime,  $\tau_2$ , over the NCs transformation. The continuous line is a fit to the data by first-order kinetics. The corresponding trend for the shorter component,  $\tau_1$ , is shown in the inset.



throughout the transformation, similar to the values measured in the solution (Fig. 5d).

The evolution of PL during the transformation was also monitored by *in situ* spectrally-resolved transient PL. The temporal PL decay is sub-ns and contains two main components, the shorter ( $\sim 70$  ps) and longer (950 ps) ones (Fig. 5e and S47†). The shorter decay component varies little over the course of the transformation while the longer decay component decreases from  $\sim 1.1$  ns to  $\sim 800$  ps with a time constant of  $\sim 10$  minutes (Fig. 5f). The PL decay of the emitting NCs in the film is much shorter than that of the NCs in solution ( $\sim 4$ – $5$  ns), the polymer-free CsPbBr<sub>3</sub> NCs<sup>5,35,56–58</sup> ( $\sim 2$ – $10$  ns), and polymer-encapsulated single CsPbBr<sub>3</sub> NCs ( $\sim 6$  ns).<sup>18</sup> It is definitely much shorter than that of MAPbBr<sub>3</sub> NCs/polymer blends ( $>100$  ns).<sup>59</sup> The fast PL decay of CsPbBr<sub>3</sub>/PMAO NCs in the drop-cast film can be attributed to various possible causes, including: (i) the appearance of a new non-radiative carrier recombination channel, ascribable to oxygen molecules (as the samples were prepared in air) which act as traps for electrons;<sup>26</sup> (ii) electron hopping between neighboring nanocrystals in the film;<sup>60</sup> (iii) a more defective surface of NCs formed in films, due to reduced mobility of ions and molecules (preventing efficient passivation of surface sites in comparison to the solution case). The sub-ns PL decay of NCs in blends with PMAO, combined with a reasonable PLQY, should be of interest for applications in scintillators, where ultrafast and efficient emission is required for fast timing capability of imaging detectors.<sup>61,62</sup>

## Conclusions

Chemical transformation of colloidal Cs<sub>4</sub>PbBr<sub>6</sub> NCs to perovskite CsPbBr<sub>3</sub> NCs induced by the organic co-polymer PMAO is presented as a promising strategy to prepare stable and bright CsPbBr<sub>3</sub> NC emitters. The PMAO reactivity towards oleylammonium/oleate-capped Cs<sub>4</sub>PbBr<sub>6</sub> NCs favors an addition reaction of oleylamine ligands from the NC surface to the succinic anhydride groups of the polymer. This destabilizes the NCs and acidifies the reaction environment through the formation of polysuccinamic acid, a PMAO–oleylamine adduct, which binds to the surface of the NCs *in lieu* of the original ligands. These two factors – ligand replacement and *in situ* acid formation – drive the Cs<sub>4</sub>PbBr<sub>6</sub> to CsPbBr<sub>3</sub> NC transformation. The lower reactivity of PMAO, as compared to that of the previously reported reagents, enabled the investigation of Cs<sub>4</sub>PbBr<sub>6</sub>–CsPbBr<sub>3</sub> intermediate heterostructures by HRTEM. The heterostructures feature a variety of epitaxial and non-epitaxial relationships between the two structurally dissimilar domains. At cryogenic temperature, Cs<sub>4</sub>PbBr<sub>6</sub>–CsPbBr<sub>3</sub> NCs display dual emission at  $\sim 376$  nm and 513 nm with evidence of energy transfer from Cs<sub>4</sub>PbBr<sub>6</sub> to CsPbBr<sub>3</sub>. The PMAO-induced transformation proceeds both in solutions and in drop-cast films, producing CsPbBr<sub>3</sub> NCs with a narrow size distribution and attractive photoluminescence properties (up to 69% PLQY in solution and a sub-ns PL lifetime in the drop-cast films). The resulting CsPbBr<sub>3</sub>/PMAO NCs demonstrate enhanced stability by retaining their green emission for several weeks in air. The increased stability of CsPbBr<sub>3</sub>/PMAO NCs is attributed to the

adhesion of polysuccinamic acid through its multiple functional groups to the NC surface. The PMAO-induced transformation of Cs<sub>4</sub>PbBr<sub>6</sub> NCs opens up a general strategy for chemical modification of inorganic NCs passivated with nucleophilic amines.

## Author contributions

The manuscript was written through the contributions of all authors. All authors have given approval to the final version of the manuscript.

## Conflicts of interest

The authors declare no competing financial interest.

## Abbreviations

COD	Crystallography open database
EDS	Energy dispersive X-ray spectroscopy
FFT	Fast Fourier transform
FTIR	Fourier transform infrared spectroscopy
HRTEM	High resolution TEM
HSQC	Heteronuclear single quantum coherence
ICSD	Inorganic crystal structure database
NIR	Near infrared
NMR	Nuclear magnetic resonance
NC	Nanocrystal
PL	Photoluminescence
PMAO	Poly(maleic anhydride- <i>alt</i> -1-octadecene)
QY	Quantum yield
STEM	Scanning TEM
TEM	Transmission electron microscopy
XRD	X-ray diffraction

## Acknowledgements

We thank Simone Lauciello and Dr Rosaria Brescia (IIT Electron Microscopy Facility) for the assistance with EDS measurements; Mr Aniruddha Ray and Dr Ahmed Abdelhady for providing samples of bulk Cs<sub>4</sub>PbBr<sub>6</sub> powders for the control experiments; Dr Urko Petralanda, Dr Ivan Infante, and Mr Stefano Toso for helpful discussions; Dr Luana Persano for support in optical measurements and film sample preparation. The work of Dmitry Baranov was supported by the European Union's Horizon 2020 Research and Innovation Programme under Marie Skłodowska-Curie grant agreement No 794560 (RETAIN). Liberato Manna acknowledges funding from the European Union under grant agreement No 614897 (ERC Grant TRANS-NANO). Riccardo Scarfiello acknowledges financial support by the Progetto FISR – C.N.R., “Tecnopolo di nanotecnologia e fotonica per la medicina di precisione” – CUP B83B17000010001. Andrea Camposo and Dario Pisignano acknowledge funding from the European Research Council under the European Union's Horizon 2020 Research and Innovation Programme (Grant Agreement n. 682157,



“xPRINT”), and from MIUR (project “3D-Phys”, PRIN 2017PHRM8X).

## References

- 1 M. V. Kovalenko, L. Protesescu and M. I. Bodnarchuk, Properties and potential optoelectronic applications of lead halide perovskite nanocrystals, *Science*, 2017, **358**(6364), 745–750.
- 2 Q. A. Akkerman, G. Rainò, M. V. Kovalenko and L. Manna, Genesis, challenges and opportunities for colloidal lead halide perovskite nanocrystals, *Nat. Mater.*, 2018, **17**(5), 394–405.
- 3 J. Shamsi, A. S. Urban, M. Imran, L. De Trizio and L. Manna, Metal Halide Perovskite Nanocrystals: Synthesis, Post-Synthesis Modifications, and Their Optical Properties, *Chem. Rev.*, 2019, **119**(5), 3296–3348.
- 4 G. Nedelcu, L. Protesescu, S. Yakunin, M. I. Bodnarchuk, M. J. Grotevent and M. V. Kovalenko, Fast Anion-Exchange in Highly Luminescent Nanocrystals of Cesium Lead Halide Perovskites (CsPbX<sub>3</sub>, X = Cl, Br, I), *Nano Lett.*, 2015, **15**(8), 5635–5640.
- 5 L. Protesescu, S. Yakunin, M. I. Bodnarchuk, F. Krieg, R. Caputo, C. H. Hendon, R. X. Yang, A. Walsh and M. V. Kovalenko, Nanocrystals of Cesium Lead Halide Perovskites (CsPbX<sub>3</sub>, X = Cl, Br, and I): Novel Optoelectronic Materials Showing Bright Emission with Wide Color Gamut, *Nano Lett.*, 2015, **15**(6), 3692–3696.
- 6 Q. A. Akkerman, V. D’Innocenzo, S. Accornero, A. Scarpellini, A. Petrozza, M. Prato and L. Manna, Tuning the Optical Properties of Cesium Lead Halide Perovskite Nanocrystals by Anion Exchange Reactions, *J. Am. Chem. Soc.*, 2015, **137**(32), 10276–10281.
- 7 Q. A. Akkerman, S. Park, E. Radicchi, F. Nunzi, E. Mosconi, F. De Angelis, R. Brescia, P. Rastogi, M. Prato and L. Manna, Nearly Monodisperse Insulator Cs<sub>4</sub>PbX<sub>6</sub> (X = Cl, Br, I) Nanocrystals, Their Mixed Halide Compositions, and Their Transformation into CsPbX<sub>3</sub> Nanocrystals, *Nano Lett.*, 2017, **17**(3), 1924–1930.
- 8 Z. Liu, Y. Bekenstein, X. Ye, S. C. Nguyen, J. Swabeck, D. Zhang, S.-T. Lee, P. Yang, W. Ma and A. P. Alivisatos, Ligand Mediated Transformation of Cesium Lead Bromide Perovskite Nanocrystals to Lead Depleted Cs<sub>4</sub>PbBr<sub>6</sub> Nanocrystals, *J. Am. Chem. Soc.*, 2017, **139**(15), 5309–5312.
- 9 F. Palazon, C. Urso, L. De Trizio, Q. Akkerman, S. Marras, F. Locardi, I. Nelli, M. Ferretti, M. Prato and L. Manna, Postsynthesis Transformation of Insulating Cs<sub>4</sub>PbBr<sub>6</sub> Nanocrystals into Bright Perovskite CsPbBr<sub>3</sub> through Physical and Chemical Extraction of CsBr, *ACS Energy Lett.*, 2017, **2**(10), 2445–2448.
- 10 T. Udayabhaskararao, L. Houben, H. Cohen, M. Menahem, I. Pinkas, L. Avram, T. Wolf, A. Teitelboim, M. Leskes, O. Yaffe, D. Oron and M. Kazes, A Mechanistic Study of Phase Transformation in Perovskite Nanocrystals Driven by Ligand Passivation, *Chem. Mater.*, 2018, **30**(1), 84–93.
- 11 Y. Li, H. Huang, Y. Xiong, S. V. Kershaw and A. L. Rogach, Reversible transformation between CsPbBr<sub>3</sub> and Cs<sub>4</sub>PbBr<sub>6</sub> nanocrystals, *CrystEngComm*, 2018, **20**(34), 4900–4904.
- 12 L. Wu, H. Hu, Y. Xu, S. Jiang, M. Chen, Q. Zhong, D. Yang, Q. Liu, Y. Zhao, B. Sun, Q. Zhang and Y. Yin, From Nonluminescent Cs<sub>4</sub>PbX<sub>6</sub> (X = Cl, Br, I) Nanocrystals to Highly Luminescent CsPbX<sub>3</sub> Nanocrystals: Water-Triggered Transformation through a CsX-Stripping Mechanism, *Nano Lett.*, 2017, **17**(9), 5799–5804.
- 13 H. Hu, L. Wu, Y. Tan, Q. Zhong, M. Chen, Y. Qiu, D. Yang, B. Sun, Q. Zhang and Y. Yin, Interfacial Synthesis of Highly Stable CsPbX<sub>3</sub>/Oxide Janus Nanoparticles, *J. Am. Chem. Soc.*, 2018, **140**(1), 406–412.
- 14 L. Yang, T. Wang, Q. Min, B. Liu, Z. Liu, X. Fan, J. Qiu, X. Xu, J. Yu and X. Yu, High Water Resistance of Monoclinic CsPbBr<sub>3</sub> Nanocrystals Derived from Zero-Dimensional Cesium Lead Halide Perovskites, *ACS Omega*, 2019, **4**(3), 6084–6091.
- 15 M. Chen, H. Hu, Y. Tan, N. Yao, Q. Zhong, B. Sun, M. Cao, Q. Zhang and Y. Yin, Controlled growth of dodecapod-branched CsPbBr<sub>3</sub> nanocrystals and their application in white light emitting diodes, *Nano Energy*, 2018, **53**, 559–566.
- 16 Y. Wei, Z. Cheng and J. Lin, An overview on enhancing the stability of lead halide perovskite quantum dots and their applications in phosphor-converted LEDs, *Chem. Soc. Rev.*, 2019, **48**(1), 310–350.
- 17 S. N. Raja, Y. Bekenstein, M. A. Koc, S. Fischer, D. Zhang, L. Lin, R. O. Ritchie, P. Yang and A. P. Alivisatos, Encapsulation of Perovskite Nanocrystals into Macroscale Polymer Matrices: Enhanced Stability and Polarization, *ACS Appl. Mater. Interfaces*, 2016, **8**(51), 35523–35533.
- 18 G. Rainò, A. Landuyt, F. Krieg, C. Bernasconi, S. T. Ochsenbein, D. N. Dirin, M. I. Bodnarchuk and M. V. Kovalenko, Underestimated Effect of a Polymer Matrix on the Light Emission of Single CsPbBr<sub>3</sub> Nanocrystals, *Nano Lett.*, 2019, **19**(6), 3648–3653.
- 19 Y. Wang, Y. Zhu, J. Huang, J. Cai, J. Zhu, X. Yang, J. Shen, H. Jiang and C. Li, CsPbBr<sub>3</sub> Perovskite Quantum Dots-Based Monolithic Electrospun Fiber Membrane as an Ultraprecise and Ultrasensitive Fluorescent Sensor in Aqueous Medium, *J. Phys. Chem. Lett.*, 2016, **7**(21), 4253–4258.
- 20 H. Kim, S. So, A. Ribbe, Y. Liu, W. Hu, V. V. Duzhko, R. C. Hayward and T. Emrick, Functional polymers for growth and stabilization of CsPbBr<sub>3</sub> perovskite nanoparticles, *Chem. Commun.*, 2019, **55**(12), 1833–1836.
- 21 H. Kim, N. Hight-Huf, J.-H. Kang, P. Bisnoff, S. Sundararajan, T. Thompson, M. Barnes, R. Hayward and T. S. Emrick, Polymer Zwitterions for Stabilization of CsPbBr<sub>3</sub> Perovskite Nanoparticle and Nanocomposite Films, *Angew. Chem., Int. Ed.*, DOI: 10.1002/anie.201916492, accepted article.
- 22 T. Pellegrino, L. Manna, S. Kudera, T. Liedl, D. Koktysh, A. L. Rogach, S. Keller, J. Rädler, G. Natile and W. J. Parak, Hydrophobic Nanocrystals Coated with an Amphiphilic Polymer Shell: A General Route to Water Soluble Nanocrystals, *Nano Lett.*, 2004, **4**(4), 703–707.



- 23 C.-A. J. Lin, R. A. Sperling, J. K. Li, T.-Y. Yang, P.-Y. Li, M. Zanella, W. H. Chang and W. J. Parak, Design of an Amphiphilic Polymer for Nanoparticle Coating and Functionalization, *Small*, 2008, **4**(3), 334–341.
- 24 R. Di Corato, A. Quarta, P. Piacenza, A. Ragusa, A. Figuerola, R. Buonsanti, R. Cingolani, L. Manna and T. Pellegrino, Water solubilization of hydrophobic nanocrystals by means of poly(maleic anhydride-*alt*-1-octadecene), *J. Mater. Chem.*, 2008, **18**(17), 1991–1996.
- 25 G. Almeida, L. Goldoni, Q. Akkerman, Z. Dang, A. H. Khan, S. Marras, I. Moreels and L. Manna, Role of Acid–Base Equilibria in the Size, Shape, and Phase Control of Cesium Lead Bromide Nanocrystals, *ACS Nano*, 2018, **12**(2), 1704–1711.
- 26 C. Rodà, A. L. Abdelhady, J. Shamsi, M. Lorenzon, V. Pinchetti, M. Gandini, F. Meinardi, L. Manna and S. Brovelli, O<sub>2</sub> as a molecular probe for nonradiative surface defects in CsPbBr<sub>3</sub> perovskite nanostructures and single crystals, *Nanoscale*, 2019, **11**(16), 7613–7623.
- 27 M. Velázquez, A. Ferrier, S. Péchev, P. Gravereau, J.-P. Chaminade, X. Portier and R. Moncorgé, Growth and characterization of pure and Pr<sup>3+</sup>-doped Cs<sub>4</sub>PbBr<sub>6</sub> crystals, *J. Cryst. Growth*, 2008, **310**(24), 5458–5463.
- 28 C. C. Stoumpos, C. D. Malliakas, J. A. Peters, Z. Liu, M. Sebastian, J. Im, T. C. Chasapis, A. C. Wibowo, D. Y. Chung, A. J. Freeman, B. W. Wessels and M. G. Kanatzidis, Crystal Growth of the Perovskite Semiconductor CsPbBr<sub>3</sub>: A New Material for High-Energy Radiation Detection, *Cryst. Growth Des.*, 2013, **13**(7), 2722–2727.
- 29 R. Kluger and J. C. Hunt, Aminolysis of maleic anhydride. Kinetics and thermodynamics of amide formation, *J. Am. Chem. Soc.*, 1984, **106**(19), 5667–5670.
- 30 R. Kluger and J. C. Hunt, Circumventive catalysis: contrasting reaction patterns of tertiary and primary amines with cyclic anhydrides and the avoidance of intermediates, *J. Am. Chem. Soc.*, 1989, **111**(9), 3325–3328.
- 31 Z. Jin, L. Du, C. Zhang, Y. Sugiyama, W. Wang, G. Palui, S. Wang and H. Mattoussi, Modification of Poly(maleic anhydride)-Based Polymers with H<sub>2</sub>N-R Nucleophiles: Addition or Substitution Reaction?, *Bioconjugate Chem.*, 2019, **30**(3), 871–880.
- 32 L. Coleman, J. Bork and H. Dunn, Notes. Reaction of Primary Aliphatic Amines with Maleic Anhydride, *J. Org. Chem.*, 1959, **24**(1), 135–136.
- 33 I. Vermeesch and G. Groeninckx, Chemical modification of poly(styrene-co-maleic anhydride) with primary N-alkylamines by reactive extrusion, *J. Appl. Polym. Sci.*, 1994, **53**(10), 1365–1373.
- 34 J. De Roo, M. Ibáñez, P. Geiregat, G. Nedelcu, W. Walravens, J. Maes, J. C. Martins, I. Van Driessche, M. V. Kovalenko and Z. Hens, Highly Dynamic Ligand Binding and Light Absorption Coefficient of Cesium Lead Bromide Perovskite Nanocrystals, *ACS Nano*, 2016, **10**(2), 2071–2081.
- 35 M. I. Bodnarchuk, S. C. Boehme, S. ten Brinck, C. Bernasconi, Y. Shynkarenko, F. Krieg, R. Widmer, B. Aeschlimann, D. Günther, M. V. Kovalenko and I. Infante, Rationalizing and Controlling the Surface Structure and Electronic Passivation of Cesium Lead Halide Nanocrystals, *ACS Energy Lett.*, 2019, **4**(1), 63–74.
- 36 D. Quarta, M. Imran, A.-L. Capodilupo, U. Petralanda, B. van Beek, F. De Angelis, L. Manna, I. Infante, L. De Trizio and C. Giansante, Stable Ligand Coordination at the Surface of Colloidal CsPbBr<sub>3</sub> Nanocrystals, *J. Phys. Chem. Lett.*, 2019, **10**(13), 3715–3726.
- 37 S. Watanabe, H. Kawahara and T. Kuramochi, Adducts of cyclic acid anhydrides and fatty amines as anti-rust additives in water-based cutting fluids, *J. Am. Oil Chem. Soc.*, 1991, **68**(2), 92–94.
- 38 M. Abbas and C. Slugovec, Optimized reaction conditions for the cross-metathesis of methyl oleate and oleylamine with ethyl acrylate, *Monatsh. Chem.*, 2012, **143**(4), 669–673.
- 39 S. Percec, L. Howe, J. Li and S. Bair, Chemical modification of poly(ethylene-co-methyl acrylate-co-maleic anhydride) for cathodic electrodepositions, *J. Polym. Sci., Part A: Polym. Chem.*, 2012, **50**(2), 261–270.
- 40 C. Zhang, C. Gao, F. Gao, J. Wang, D. Zhang, Y. Wang and D. Xu, Synthesis of comb copolymers and their pour point depressing properties, *Pet. Sci.*, 2014, **11**(1), 155–160.
- 41 G. Almeida, O. J. Ashton, L. Goldoni, D. Maggioni, U. Petralanda, N. Mishra, Q. A. Akkerman, I. Infante, H. J. Snaith and L. Manna, The Phosphine Oxide Route toward Lead Halide Perovskite Nanocrystals, *J. Am. Chem. Soc.*, 2018, **140**(44), 14878–14886.
- 42 S. Park, N. M. An, G. Almeida, F. Palazon, D. Spirito, R. Krahne, Z. Dang, L. De Trizio and L. Manna, CsPbX<sub>3</sub>/SiO<sub>x</sub> (X=Cl, Br, I) Monoliths Prepared via a Novel Sol-gel Route Starting from Cs<sub>4</sub>PbX<sub>6</sub> Nanocrystals, *Nanoscale*, 2019, **11**(40), 18739–18745.
- 43 Y. Liu, Z. Wang, S. Liang, Z. Li, M. Zhang, H. Li and Z. Lin, Polar Organic Solvent-Tolerant Perovskite Nanocrystals Permanently Ligated with Polymer Hairs via Star-like Molecular Bottlebrush Trilobe Nanoreactors, *Nano Lett.*, 2019, **19**(12), 9019–9028.
- 44 M. Meyns, M. Perálvarez, A. Heuer-Jungemann, W. Hertog, M. Ibáñez, R. Nafria, A. Genç, J. Arbiol, M. V. Kovalenko, J. Carreras, A. Cabot and A. G. Kanaras, Polymer-Enhanced Stability of Inorganic Perovskite Nanocrystals and Their Application in Color Conversion LEDs, *ACS Appl. Mater. Interfaces*, 2016, **8**(30), 19579–19586.
- 45 H. Wu, S. Wang, F. Cao, J. Zhou, Q. Wu, H. Wang, X. Li, L. Yin and X. Yang, Ultrastable Inorganic Perovskite Nanocrystals Coated with a Thick Long-Chain Polymer for Efficient White Light-Emitting Diodes, *Chem. Mater.*, 2019, **31**(6), 1936–1940.
- 46 C. Carrillo-Carrión, P. del Pino and B. Pelaz, Aqueous stable luminescent perovskite-polymer composites, *Applied Materials Today*, 2019, **15**, 562–569.
- 47 K. Momma and F. Izumi, VESTA 3 for three-dimensional visualization of crystal, volumetric and morphology data, *J. Appl. Crystallogr.*, 2011, **44**(6), 1272–1276.
- 48 A. D. Wright, C. Verdi, R. L. Milot, G. E. Eperon, M. A. Pérez-Osorio, H. J. Snaith, F. Giustino, M. B. Johnston and



- L. M. Herz, Electron–phonon coupling in hybrid lead halide perovskites, *Nat. Commun.*, 2016, **7**(1), 11755.
- 49 Y. Guo, O. Yaffe, T. D. Hull, J. S. Owen, D. R. Reichman and L. E. Brus, Dynamic emission Stokes shift and liquid-like dielectric solvation of band edge carriers in lead-halide perovskites, *Nat. Commun.*, 2019, **10**(1), 1175.
- 50 M. Nikl, E. Mihokova, K. Nitsch, F. Somma, C. Giampaolo, G. P. Pazzi, P. Fabeni and S. Zazubovich, Photoluminescence of Cs<sub>4</sub>PbBr<sub>6</sub> crystals and thin films, *Chem. Phys. Lett.*, 1999, **306**(5), 280–284.
- 51 V. Babin, P. Fabeni, E. Mihokova, M. Nikl, G. P. Pazzi, N. Zazubovich and S. Zazubovich, Luminescence of Cs<sub>4</sub>PbBr<sub>6</sub> Aggregates in As-Grown and in Annealed CsBr:Pb Single Crystals, *Phys. Status Solidi B*, 2000, **219**(1), 205–214.
- 52 S. Radhakrishna and K. P. Pande, Lead Centers in Cesium Halides, *Phys. Rev. B*, 1973, **7**(1), 424–431.
- 53 P. W. M. Jacobs, Alkali halide crystals containing impurity ions with the ns<sup>2</sup> ground-state electronic configuration, *J. Phys. Chem. Solids*, 1991, **52**(1), 35–67.
- 54 J. Yin, Y. Zhang, A. Bruno, C. Soci, O. M. Bakr, J.-L. Brédas and O. F. Mohammed, Intrinsic Lead Ion Emissions in Zero-Dimensional Cs<sub>4</sub>PbBr<sub>6</sub> Nanocrystals, *ACS Energy Lett.*, 2017, **2**(12), 2805–2811.
- 55 J. Hoy, P. J. Morrison, L. K. Steinberg, W. E. Buhro and R. A. Loomis, Excitation Energy Dependence of the Photoluminescence Quantum Yields of Core and Core/Shell Quantum Dots, *J. Phys. Chem. Lett.*, 2013, **4**(12), 2053–2060.
- 56 M. Imran, V. Caligiuri, M. Wang, L. Goldoni, M. Prato, R. Krahn, L. De Trizio and L. Manna, Benzoyl Halides as Alternative Precursors for the Colloidal Synthesis of Lead-Based Halide Perovskite Nanocrystals, *J. Am. Chem. Soc.*, 2018, **140**(7), 2656–2664.
- 57 M. Imran, P. Ijaz, D. Baranov, L. Goldoni, U. Petralanda, Q. Akkerman, A. L. Abdelhady, M. Prato, P. Bianchini, I. Infante and L. Manna, Shape-Pure, Nearly Monodispersed CsPbBr<sub>3</sub> Nanocubes Prepared Using Secondary Aliphatic Amines, *Nano Lett.*, 2018, **18**(12), 7822–7831.
- 58 M. Imran, P. Ijaz, L. Goldoni, D. Maggioni, U. Petralanda, M. Prato, G. Almeida, I. Infante and L. Manna, Simultaneous Cationic and Anionic Ligand Exchange For Colloidally Stable CsPbBr<sub>3</sub> Nanocrystals, *ACS Energy Lett.*, 2019, **4**(4), 819–824.
- 59 Y. Wang, J. He, H. Chen, J. Chen, R. Zhu, P. Ma, A. Towers, Y. Lin, A. J. Gesquiere, S.-T. Wu and Y. Dong, Ultrastable, Highly Luminescent Organic–Inorganic Perovskite–Polymer Composite Films, *Adv. Mater.*, 2016, **28**(48), 10710–10717.
- 60 S. J. Yoon, Z. Guo, P. C. dos Santos Claro, E. V. Shevchenko and L. Huang, Direct Imaging of Long-Range Exciton Transport in Quantum Dot Superlattices by Ultrafast Microscopy, *ACS Nano*, 2016, **10**(7), 7208–7215.
- 61 C. Dujardin, E. Auffray, E. Bourret-Courchesne, P. Dorenbos, P. Lecoq, M. Nikl, A. N. Vasil'ev, A. Yoshikawa and R. Zhu, Needs, Trends, and Advances in Inorganic Scintillators, *IEEE Trans. Nucl. Sci.*, 2018, **65**(8), 1977–1997.
- 62 K. Tomanová, V. Čuba, M. G. Brik, E. Mihóková, R. M. Turtos, P. Lecoq, E. Auffray and M. Nikl, On the structure, synthesis, and characterization of ultrafast blue-emitting CsPbBr<sub>3</sub> nanoplatelets, *APL Mater.*, 2019, **7**(1), 011104.

