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Extraction and modification of cellulose nanofibers derived from biomass for environmental application

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Cellulose is a natural biopolymer that is abundantly available in plant cell walls and is secreted in its pure forms by many bacteria. Due to their unique features cellulose materials are considered as efficient replacements for conventional polymers. Cellulose nanofibers (CNF) have attracted wide interest due to their nano size, ease of preparation, low cost, tuneable surface properties and enhanced mechanical properties. However, the efficiency of CNF depends on the extraction method employed from its source and their features vary from source to source. Hence, there is a need to understand the specificity of CNF extraction from its source in order to obtain highly efficient CNF with maximum potential. CNF has been extracted from plant sources using physical, chemical and enzymatic methods. Although plant derived CNF possess excellent features, the involvement of chemicals and complexity in extraction process limits their usage. Bacterial CNF overcome this limitation through its extracellular secretion which makes extraction easy. CNF is also extracted from various marine filamentous algae. The percentage of CNF obtained from algal sources is less compared to plants and bacterial sources. CNF finds wide variety of applications such as drug carriers, tissue regenerating scaffolds, water purifying

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membranes, electrodes, supercapacitors, fluorescent probes and flexible electronics. In this review, various extraction techniques of CNF from different plant and bacterial sources are discussed critically with special emphasis on CNF based composites.

1. Introduction

Cellulose (chemical formula: $(C_6H_{10}O_5)_n$), the most abundant renewable polysaccharides in plants and microorganisms is composed of repeating units of β-D-glucopyranose bound through covalent linkage between the OH group of C4 and C1 carbon atoms.[1,2] Based on its source, the obtained cellulose exhibits a characteristic structural hierarchy that can be interrupted using suitable extraction procedures to obtain various forms of cellulose.[3,4] For more information about the details of structure, source, properties and various forms of cellulose, please refer to Moon et al.[5] The properties of natural cellulose structures like shape, length and diameter depend on its origin and the extraction process. Hence variations in cellulosic properties are more likely and need individual study on cellulose derived from different plants.[6] Despite various advantages, natural cellulose fibres exhibit few limitations that restrict its widespread applications such as poor thermal stability, non-compatibility with hydrophobic polymers, absorption of moisture etc.[7,8] Due to these limitations, there is a requirement of converting readily available cellulose into nano/micro-cellulosic forms or blending it with appropriate polymers to form composites which can enhance their desired properties. When compared to cellulosic microfibrils, nanoform of cellulose has high surface-to-volume value and nano-scale size effect that makes it superior.[9] Cellulose extracted from plants[3] and

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secreted by microorganism[10] have been converted into various nanostructures like nanofibers,[11] hydrogels,[10] nanoparticles,[5] aerogels,[12] films,[5] nanocrystals,[13] nanowhiskers[14] etc. Among the various nanostructures derived from cellulose, cellulosic nanofibers have been widely investigated due to its excellent mechanical strength, flexibility, possibility of functionalization and blending, biodegradability, chirality, thermostability and low thermal expansion. Cellulose nanofibers (CNF) has been considered as potential matrix for various applications like optoelectronic conversion, energy storage, packaging, drug delivery, bioimaging and biomedical materials, nanofillers, protective coatings, barrier membranes and filtration media, transparent films, antimicrobial activity, pharmaceuticals etc.[5,15,16] CNFs are renewable, low cost and low density material with less abrasive property.[17]

The properties of CNF such as purity, biocompatibility, crystallinity, wettability and surface tuneable structure have intense effect on various environmental applications. Biocompatibility of CNF is an important property that makes them highly suitable for various biomedical applications especially in scaffold development for tissue engineering. This is also an important property for its environmental applications.[18] As filtration system or membranes for water purification, the CNF should not degrade and release toxic materials that make the treated water unfit for human consumption. Highly pure CNFs are important in environmental aspect. This is due to the fact that, increase in CNF purity enhances the thermal stability of the fibers. This is especially important in applications such as sensor fabrication and catalytic decomposition of various pollutants in waste water remediation where thermal stability of CNFs is highly demanded. In catalytic decomposition, the CNF acts as catalyst and prevent the aggregation of nanoparticles such as TiO_2, Au and CuO that are introduced along with CNF for decomposing the pollutants. High purity of CNFs also improves the crystallinity of the nanofibers. This enhances the compactness of the CNF structure thereby maintaining their integrity. Highly crystalline nature increases the stability of the CNF structure. When combined with other materials having low crystallinity such as PVA, PLA etc., CNF imparts stiffness and high strength to the nanocomposite in addition to the native properties of the added materials.[19,20] The high crystallinity observed in CNF also enhances their sorption property towards various volatile compounds from water and air. Thus, CNF based adsorbents can be used in the remediation of air and water pollution. Besides this, the sorption ability of CNF can also be utilised in sensors, catalysis and molecular level compound separation applications. CNF can accommodate a wide variety of guest nanoparticles such as Au, Ag, Pb, Ni, TiO_2, CuO, SiO_2 etc. in its structure. This is due to their high mechanical stability, intact crystalline structure, surface area and porosity. An important feature of CNFs is the high concentration of reactive –OH groups present on their surface that helps in CNF modification. These OH groups present on the cellulose surface crosslinks with quaternary ammonium compounds forming quaternary amines. This process is called quaternization. Through this process, cationic groups are introduced on the CNF surface to improve the adsorption of

anions from polluted water sources.[21] Apart from being used as ion exchange resins for water purification, the tuneable surface property of CNFs is also exploited in the fabrication of anti-bacterial water filters, energy devices and sensor applications. For example, CNF–AgNp based water filters prevents biofouling of the membrane and improves the life of the filter used.[22] The high wettability of CNFs is used in various environmental applications such as water purification, development of absorbents for oil–water separation from oil-spills, anti-fouling coatings etc. The wettability of CNF plays an important role in water and oil separation. While conventional methods have drawbacks of ineffectiveness, high demand of chemicals and energy, CNF based adsorbents (sponges and filters) are effective and efficient alternatives. The CNF adsorbents allow selective permeation of either oil/water and omit the other compound. CNF based environmental friendly anti-fouling coatings has been developed to protect the surfaces of marine vessels, environmental sensors, water treatment plants etc. The CNF coatings due to high wettability prevent the attachment and growth of the microbial flora on the surfaces.

The CNF's can be prepared using physical, chemical, biological and oxidation methods.[23] In chemical methods, cellulosic nanofibers are prepared through acid digestion, where amorphous area of fibers is destroyed to yield nanocrystalline nanofibers.[24] During physical methods, the wood pulps are ground at high rpm, subjected to high intensity ultrasonic treatment, mechanical nanofibrillation[25] to obtain nanofibers of nanometer diameter.[26] In case of biological treatments, cellulosic materials are treated with cellulolytic enzymes like cellulase that cleave the fiber structures to simpler ones.[15,26] Since biological treatment takes more time, they are often coupled with mechanical/chemical methods to reduce the process time and get better CNF.[23] Better yield of CNF has been obtained through oxidation of CNFs mediated through 2,2,6,6-tetrame-thylpiperidine-1-oxyl radical.[27] The exposure of such oxidised fibers to mechanical treatment results in easier defibrillation due to the generation of negative charges that repels the microfibrils against each other inside the cell wall.[27] The main purpose of the oxidation process is to make the secondary

Table 1   Percentage of cellulose present in various plant sources

Source	Cellulose content (%)	Reference
Rice husk		
Ahu variety	94.2	87
Boro variety	89.6	66
Banana	16.62	60
Eucalyptus	76	93
Wheat straw	43	89
Rice straw	71	130
Flax fibers	72	153
Cotton fibers	85–90	154
Corn silage	32	155
Hemp fibers	68	155
Sisal fibers	65.5	156
Cassava bran	16.71	27
Achirafibers	19.1	50

Table 2 Shows various sources of CNF, their preparation techniques and their properties

Source of CNF	Method of preparation	Properties of the material developed	Ref.
Natural cellulose			
Pineapple peel juice, *Gluconacetobacter swingsii* sp.	Spray coating	Anti-microbial activity, enhanced spread factor ($\xi_{max} = +0.040C^2$ and $-0.075C^2$)	136
Rice husk from *Oryza sativa*	Hydrothermal approach, acid-alkali treatment, mechanical disruption	Size; 30–40 nm, innate fluorescence property, purity, crystallinity, thermostability	87
Fibrous residues of *Achira rhizomes*	Acid hydrolysis, high pressure homogenisation	Size; 13.8–37.2 nm, high crystallinity ($I_{cr} = 57.5\%$ and 69.8%), biodegradability, mechanical stability	50
Banana peel	Chemical and enzymatic treatment using xylanase	Size; 10.9 nm and 7.6 nm, biodegradability, high crystallinity ($I_{cr} = 49.2\%$)	64
Powder from poplar wood	Chemical pretreatment, high intensity ultrasonication	Size; 5–20 nm, high thermostability (335 °C), high crystallinity (69.34%)	61
Poplar wood, culms of moso bamboo, rice straw, corn straw	Chemical treatment, ultrasonication, high pressure homogenisation	Size; 2–5 nm, high stability, ribbon like structure, high flexibility	25
Tomato peels	Acidified sodium chlorite, chlorine free alkaline peroxide	Size; 260 ± 79 nm, high crystallinity ($I_{cr} = 69\%$)	146
Posidonia oceanica balls and leaves	Chemical treatment, fibrillation	Size; 5–21 nm and 2–15 nm	53
Cotton stalks	Chemical treatment, ultrasonication, mechanical treatment	Size; 3–15 nm, cost effective, biodegradable	17
Waste pulp residues from paper industry	Etherification of pulp, mechanical disintegration	Size; 10–100 nm, high fibrillation, high thermostability (320 °C), high nitrate adsorption capacity (0.7 mmol g^{-1})	92
Culinary banana peel	Chemical treatment, high intensity ultrasonication	High crystallinity ($I_{cr} = 63.64\%$), high thermal stability (295.33 °C)	60
Eucalyptus pulp	TEMPO mediated oxidation	High water retention value (WRV = 8.3 g g^{-1}), high modulus of rupture and modulus of elasticity (MOR = 35 MPa, MOE = 5160 MPa), high strength	93
Canola straw	Nanowelding	Size; 53 ± 16 nm, high tensile strength (208 MPa), Young's modulus (20 GPa), superior transparency (76%), biodegradability	147
Oil palm trunk, oil palm frond, okara	Alkaline treatment, electrospinning	Size; <500 nm, high fiber content (107.9%, 67.2%, 25.1%), high anti-oxidant activity (377.2%, 367.8%), superior mineral (Fe, Zn, Cu, Ca) binding activity, high emulsion activity ($66.3 \pm 0.6\%$, $6.6 \pm 0.1\%$, $4.0 \pm 0.1\%$)	85
Bacterial cellulose (native)			
Acetobacter xylinum (ATCC10245)	Static culture	High cytochrome c adsorption efficiency (36.4 mg g^{-1}), high protein binding efficiency, high selectivity	148
Acetobacter xylinum X-2	Static culture	Size; 68 ± 15 nm and 117 ± 15 nm, enhanced cell proliferation, high crystallinity ($I_{cr} = 88.3\%$), excellent hydrophilicity with contact angle $50 \pm 2.4°$, favourable thermal stability (290–370 °C), biocompatibility	131
Gluconoacetobacter hansenii (strain NCIM 2529)	Static culture	BSA protein adsorption (>90%), high bioadsorption of Pb^{2+}, enhances porosity and water holding capacity of soil	115
Acetobacter xylinum FF-88	Static culture obtained from (Fujicco Co., Ltd.)	High flexibility, high transparency (90%), low coefficient of thermal expansion (4 ppm K^{-1}), low thermal expansion (0.05%), high mechanical properties	115
Acetobacter xylinum ATCC-700178	Static culture	Cost effective, biocompatibility, promotes migration of fibroblast cells, enhanced deposition of collagen, assist wound closure	149
Acetobacter xylinum (subspecies-sucrofermentas BPR2001, Trade number 700178™)	Static culture	Enhanced porosity, high Young's modulus (8.25(1.14) MPa), biodegradable, enhanced migration of smooth muscle cells	101
Synthetic cellulose (cellulose acetate)			
Commercially obtained	Electrospinning	Size; 10 µm, biocompatibility	144
Commercially obtained	Electrospinning	Size; 0.59 ± 0.24 µm, cost effective, exhibited reverse phase behaviour	145

Table 2 (Contd.)

Source of CNF	Method of preparation	Properties of the material developed	Ref.
Commercially obtained	Electrospinning	Size; 300 nm to 1.5 μm, high specific surface area (4.39 m 23 g^{-1})	150
Commercially obtained	Electrospinning	Size; 450 nm, enhanced blending capacity with polymers, enhanced thermostability	151
Commercially obtained	Electrospinning	Size; 4.6 ± 1.8 μm and 8.1 ± 2.2 μm, high fluid permeability (8.9 × 10^{-12} m^2)	152

cell wall accessible to mechanical treatment by loosening the primary cell wall. Besides, the water retention value (WRV) of the CNF enhances during the oxidation process and this hydration results in swelling of the fibers thereby making the process of defibrillation easier during mechanical treatment.

Recently cellulose isolated and purified from various agriculture sources have been electrospun to get CNF of various diameters.[28] However the challenges like solubility of cellulosic biopolymer in conventional solvent system and its properties to aggregate and form gel always leaves processing problems.[29] Hence researchers have looked into extracellular/intracellular bacterial cellulose as alternative for easing the processing steps. Unlike plant cellulose, bacterial cellulose extraction does not need physical/chemical intervention. Eichhorn *et al.* reviewed various methods used for the preparation of CNF based nanocomposites.[30] The authors briefly detailed different methods employed for the extraction of CNF from plants and bacterial sources. In their work, more emphasis was given on CNF based nanocomposites preparation and their applications.[30] In another study, Chirayil *et al.* described the preparation of nanocellulose from various fibers of lignocellulosic origin. The review explains in detail various chemical and mechanical treatments employed for the extraction of nanocellulose from plant sources.[31] Similarly, Wei *et al.* in 2014 critically reviewed the environmental applications of plants and bacteria derived nanocellulose based nanocomposites. The authors focused mainly on different forms of nanocellulose such as nanofibrillated cellulose, nanocrystalline cellulose *etc.*[32] Although there are several reviews available on the CNF extraction from plants and bacterial sources, a consolidated and detailed CNF extraction procedures customised for individual plant and microbial sources, their modification and applications are not available till date. In this study, we have tried to critically review and present consolidated information on CNF extraction from individual plant (cotton, wood pulp, banana, corn and wheat straw, soy hulls and sea grasses) and microbial (bacteria and algae) sources. The review also sheds light on the development of various CNF based composites. The authors also discussed in detail about CNF functionalization using various polymers, nanoparticles and carbon for numerous applications with special emphasis on water purification, biomedical and food packaging. Since the principle behind the methods involved in production of various CNF and its advances has been reviewed already by Nechyporchuk *et al.*[33] and the information given in the review has not been included to avoid repeatability.

2. CNF derived from algal sources

Algal cellulose was first derived and described in 1885.[34] In addition to its economic advantage, the extraction of cellulose fibers from various algal sources has been considered as an environmental remediation approach.[35] The use of algae to extract cellulose fibers helps in preventing the damage caused to the marine ecosystem due to excessive and unwanted blooming of such algae.[36,37] Cellulose fibers are extracted from green filamentous algae such as *Cladophora*, *Chaetomorpha*, *Microdyction*, *Rhizoclonium* and members of *Siphonocladales*.[38] Pääkkö *et al.* reported the extraction of cellulose fibers in nanometre size from algal sources using enzymatic hydrolysis and mechanical homogenization.[39] The cellulose fibers extracted from these algae were reported to have monomers of D-glucose units that are linked by β(1–4) linkages similar to the CNFs obtained from wood sources.[40] Ek *et al.* reported that the CNF obtained from *Cladophora* has a radius of 38 nm.[41] Mihranyan *et al.* determined the specific surface area of *Cladophora* CNF to be 95 m^2 g^{-1} and was reported to have web like structure with multiple intertwined fibers. The individual fiber width was observed to be 25–30 nm.[42] The season of algal harvesting plays an important role in the strength of the CNF extracted from them. For instance, Mihranyan indicated in his study that the tensile strength of the extracted CNF from algal origin depends on the season they are harvested. Algal harvesting during the early season results in cellulose fibers with low tensile strength due to incomplete development of cell wall. Similar pattern was observed in case of fibers extracted from algae harvested towards the end of the season due to aging.[42]

Johnson *et al.* reported that midseason exhibits algal cell wall with maximum strength cellulose fibers. CNF with high tensile strength (9 Mpa) has been obtained from *Cladophora* harvested during the mid of the season.[43] The CNF extracted from filamentous marine green algae are highly crystalline in nature due to the presence of thick microfibrils of 10–30 nm width with high degree of orientation.[44,45] Algal CNFs were reported to have high density (1.64 g cm^{-3}) and less moisture absorption due to its high crystallinity. The high crystallinity in algal CNFs provides inertness that is responsible for its resistance to different chemical treatments.[46] In a study, Hayashi *et al.* reported that the algal cellulose fibers are rich in Iα allomorph that are highly susceptible to enzymatic attack.[47]

CNFs derived from algae have several industrial and environmental significance. They are used as reinforcing agents in polyurethane foams. Such cellulose fiber reinforced materials

Fig. 1 SEM images of various types of CNF synthesized from cotton (a) natural electrospun CNF[56] (b) aligned CNF (scale bar 1 μm)[54] (c): CdS functionalized CNF[33] (d): CeO$_2$ functionalized CNF[56] (e): CNF prepared by acid hydrolysis[52] (adopted with permission).

exhibits tremendous improvement in strength, elastic modulus, biodegradability and thermal resistance. *Cladophora* extracted CNFs also find immense application as filter membrane materials in water purification. They are used in drug delivery purposes due to their inert nature and high surface area. The high drug loading capacity makes them suitable carriers for liquid and solid drug materials. The pores present in the algal CNFs protect the loaded drug molecules from unfavourable environment.

Albert *et al.* reported that *Cladophora* CNFs form highly stable and transparent gels by means of high speed sonication.[48] Their strong rheological properties and robustness makes them suitable materials for food packaging, wound dressings and pharmaceutical applications. *Cladophora* CNF composites are reported as suitable materials for ion exchange resins and paper based battery fabrication. The insulating property of algal CNF is converted into conductive nature by coating the fibers with polymers such as polypyrrole (PPy) and polyaniline. Such composite algal CNFs are used for the fabrication of conductive paper based energy storage devices.[49] Such a battery cell consisting of two electrodes, PPy/*Cladophora* CNF was reported by Nyström *et al.* The algal CNF based paper batter systems are cost effective and environmental friendly.[49]

Despite all the advantages, the CNF extraction from algal sources is less compared to plants and bacterial sources. Hence, this review focusses in depth on CNFs derived from plants and bacterial sources in the following sections.

3. CNF derived from plant sources

Plant derived CNF (Table 1) has attracted researchers due to their sustainability, availability, low cost and the related characteristics such as large surface to volume ratio, high stiffness, high flexibility, good water resistance, biocompatibility, biodegradability, renewability, low density, enhanced specific strength[50] and superior thermo-mechanical properties as compared with other commercial fibers.[51] CNF has been derived from various plant sources like cotton, banana, corn, rice, aloe vera, jute, palm, soya bean, tomato peel, wood pulp and many other agro wastes. Table 1 gives an account of cellulose content present in various plant sources. Since the cellulose content in different agro wastes are varying, the method opted for synthesis of CNF also varies. The CNF from these plant sources have been synthesised using various physical, chemical and biological methods. Physical treatment involves ultra-

centrifugation, high speed homogenisation, ultra-sonication *etc.* whereas chemical treatment involves alkaline exposure, acid hydrolysis and bleaching process. In case of biological treatment process, enzymes such as cellulose and xylanase are used. Best results have been reported in treatments involving combination of these treatments. Table 2 gives detailed information on CNF isolation techniques used for their extraction from various sources, their properties and applications. The following section discusses various methods opted/modified to obtain CNF of desired quality and function.

3.1 Cotton based CNF

The cellulose content in natural white cotton (*Gossypium hirsutum* L) is almost close to 100% and varies from 74 to 90% in naturally coloured cotton. Such reduced level of cellulose in naturally coloured cotton is attributed towards the presence of more lignin and hemicellulose.[52] CNF from cotton have been prepared using various methods like acid hydrolysis, chemical/ultrasonication process/(2,2,6,6-tetramethylpiperidin-1-yl)oxy radical oxidation (TEMPO) process, electrospinning *etc.*

The preparation method, source and the surface type of CNF has considerable impact on its thermal degradation properties.[17,52] CNF prepared from white and coloured cotton by acid hydrolysis had a length of 85–225 nm and diameter of 6–18 nm. The CNF obtained from white and coloured cotton were 17 to 24 nm in size and retained its original colour in water even after acid extraction process. They also reported that the coloured CNF were thermally more stable at 180 °C than white under isothermal oxidizing conditions.[52]

The efficiency of TEMPO mediated oxidation for isolating uniform CNFs from cotton were reported by Soni *et al.*[17] They demonstrated the production of four varieties of CNF from cotton stalks using various chemical approaches. Their method of extraction comprised of fine grinding and sieving of the cotton stalks obtained using 30–80 mesh sieves. The cotton was subjected to 16 h ethanol treatment for dewaxing followed by drying at 105 °C for 18 h. The obtained purified cotton stalks were then exposed to acid/alkaline treatment for extracting the cellulose in pure form. First, the cotton stalks were treated with 15% NaOH solution for 2 h at 23 °C. After collecting the resulting fibers, they were washed well using distilled water and subjected to acid hydrolysis by treating with 1 M HCl solution for 2 h at 80 °C. After filtering and washing the acid hydrolysed fibers, they were exposed to alkaline treatment again under the conditions mentioned above. Further, the fibers were bleached with sodium chlorite solution at 75 °C followed by filtration. Soni *et al.*[17] demonstrated the extraction of CNFs from the prepared pulp using TEMPO mediated oxidation and acid hydrolysis using H$_2$SO$_4$. The advantages of including ultrasonication at the end of chemical treatments in the extraction of CNFs were also shown in their study. In acid hydrolysis, the bleached cellulose pulp was treated with 64% H$_2$SO$_4$ for 50 min at 45 °C. To this aqueous solution of Na$_2$CO$_3$ was added and subjected to centrifugation for 20 min at 9000 rpm. The centrifugation was repeated thrice and the final CNFs obtained were redispersed in water. The TEMPO mediated oxidation
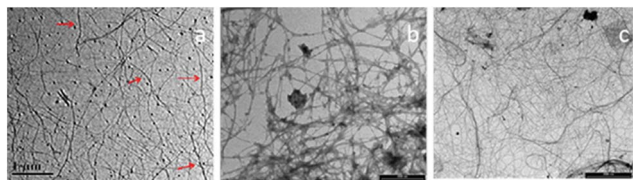
Fig. 2 TEM images of various types of CNF synthesized from banana (a): CNF after high intensity ultra-sonication;[60] (b) CNF after chemical[60] and (c) enzymatic treatment[64] (scale bar: 2 μm) (adopted with permission).

method used for isolating cellulose fibers were similar to the protocol described by Bettaieb *et al.*[53] Finally, the acid hydrolysed and TEMPO oxidised cellulose fibers were subjected to ultrasonication in ice water bath for 20 min.[17] The CNF obtained were of 3 to 50 nm in size after subjecting to H_2SO_4 treatment and TEMPO oxidation process. The thermal degradation temperature was greater with untreated bleached pulp and H_2SO_4 neutralised CNF when compared to H_2SO_4 dialysed and TEMPO oxidized CNFs. The authors suggested that the CNFs containing free sulfate group in the frame was more thermally stable in the high temperature region than that containing sodium carboxylate group, especially in oxidative conditions. He *et al.*[54] synthesized uniaxially aligned CNF (212–221 nm average dia) from cotton based cellulose nano crystals (CNC) using electrospinning process and tested it as scaffold for tissue engineering. For this purpose, a non cellulose degrading solvent system, lithium chloride/dimethyl acetamide (LiCl/DMAc) was used to electrospun non derivative cellulose onto a rotating steel drum collector. 20% loading of CNC into the cellulose solution increased the tensile strength and elastic modulus of electrospun cellulose/CNCs nanocomposite nanofibers by 101.7 and 171.6%, respectively in the fiber alignment direction. These nanocomposite nanofibers were found to be compatible for hDFCs (human dental follicle cells) attachment and proliferation. Li *et al.*[55] prepared electrospun CNF from natural cotton cellulose having degree of polymerization above 10 000. These cotton nanofibers were functionalized and made into composites by coating CeO_2 nanoparticles on surface using hydrothermal reaction. This composite nanofiber was reported to possess excellent UV-shielding properties as compared to the natural cotton CNFs that can be used for medical, military, biological, and optoelectronic applications. Similarly, Liu *et al.*[56] synthesized CdS nanoparticle-functionalized natural cotton cellulose electrospun nanofibers for photocatalytic applications.

The CNF surfaces were homogeneously dispersed with CdS nanoparticles having cubic zinc-blend structure. These heteromatrices CNF/CdS nanocomposite had excellent ability to photo degrade 99% of Rhodamine B (RhB) under visible light irradiation. The scanning electron microscopic images of various CNF synthesized by above mentioned authors are given in Fig. 1.

3.2 Banana based CNF

The cellulose content present in the banana fibers plays a major role in determining the quality of the fiber. The cellulose

content of banana fibers varies from one variety to the other and between one part to the other in the same variety. For example, the peduncle fibre of nendran variety has 60.41% cellulose whereas grand naine peduncle fibre has only 48.31% cellulose. Similarly the cellulose content in pseudostem of nendran (59.22%) and grand naine (48.19%) vary significantly (to know more about the cellulose content of various banana varieties please refer to Preethi and Murthy, 2013).[57] Hence for attaining good CNF yield, the selection of variety and the part of banana remains crucial. Natural CNF has been prepared from banana peel using high pressure hydrothermal/steam explosion process. During this process, the natural microfibers were alkali treated (2% sodium hydroxide) and subjected to steam explosion process involving pressurized steam (for *e.g.*, pressure of 0.0014 kg cm^{-2} at a temperature of 120 °C for 1 h) and oxalic acid which induces break down of lignocellulosic structure, hemicelluloses hydrolysis, depolymerisation and defibrillation of the lignin components.[58] CNF of ∼30 nm dia having increased surface acidity, thermal stability, tensile, flexural, and impact strength which can be used as reinforcement material are obtained using this method. Hooshmand *et al.*[59] synthesized filaments containing banana CNF using dry spinning method. Banana rachis was bleached and subjected to ultrafine grinding with a MKZA10 Super Masscolloider which reduced bio residue to a gel. The gel was further dry spun using a capillary rheometer to prepare continuous filaments. The dry spun CNF was collected manually on a glass sheet and dried subsequently. The filaments fracture surface and mechanical properties varied with quantity of CNF present. High-intensity ultrasonication combined with chemical treatment was used to isolate CNF having high thermal stability from culinary banana peel.[60] The process of high intensity ultrasonication transfers ultrasonic energy to the cellulose chains through cavitation process.[61] The energy provided by such cavitation process (10–100 kJ mol^{-1}) disintegrates the micron size banana cellulose fibers to individual nanosize fibers.[62] However increasing the ultrasonic energy beyond has considerable impact on nanofibrillation. Pelissari *et al.*[63] isolated CNF from banana peel using combination of alkaline treatment, bleaching, and acid hydrolysis followed by subjecting the pre-treated banana peel to high pressure homogenisation several times. The reduced particle size of CNF with increased zeta potential and stability in suspension was achieved with increase in homogenization cycle. Tibolla *et al.*[64] combined chemical and enzymatic treatment to isolate CNF from *Musa paradisiaca* (Terra) variety of plantin bananas (unriped) peel. Banana peel was subjected to alkaline treatment, bleaching, and acid hydrolysis in combination along with xylanase enzyme. In brief, the peel of the bananas were removed and soaked in 1% potassium metabisulphite solution for one day and was dried in oven at 60 °C for one day. The dried banana material was then milled and dried in oven at 60 °C for 24 h after ethanol wash. The content was sieved using 200 mesh sieve and prepared for further subjected to chemical as well as enzymatic treatments. In chemical treatment, the bran was first stirred with 5% KOH solution at room temperature (RT) for 14 h and then subjected to delignification by bleaching using 1% $NaClO_2$ for 1 h at 70 °C.

This was further subjected to second KOH treatment at RT for 14 h. The residue obtained at the end of second alkaline treatment was then exposed to acid hydrolysis using 1% H_2SO_4 solution for 1 h at 80 °C. The residues obtained at the end of each step of the acid-alkali treatment were centrifuged at 10 000 rpm for 20 min at 5 °C. The final CNF residues achieved was washed well with water and stored at 4 °C. In the enzymatic treatment, the prepared bran was pre-treated with acetate buffer and subsequently subjected to xylanase treatment for 24 h at 45 °C under constant stirring. Further, this solution was immersed in a water bath at 80 °C for 30 min. The resulting pulp was centrifuged at 10 000 rpm at 5 °C for 15 min after thorough water wash to obtain colloidal suspension of CNFs.[64] The resulted CNFs had diameters of 10.9 and 7.6 nm, when treated with chemical and enzymatic process respectively. The CNF obtained by enzymatic treatment showed stable suspension with higher zeta potential and aspect ratio when compared chemically treated banana peel. Likewise, Khawas and Deka[60] demonstrated the isolation of CNFs from banana peel by combining chemical and mechanical approach of extraction. The fresh culinary bananas (musa ABB variety) was collected and washed well using water and further soaked in 1% potassium metabisulphite solution for 12 h after separating their pulp. The peels were then subjected to tray drying for a day at 50 °C followed by grinding and sieving using 0.25 mm mesh sieve. The peel flour was first subjected to alkaline treatment for 1.5 h at 170 °C by exposing them to 20% NaOH and 0.1% anthraquinone. Followed by distilled water wash, they were further subjected to bleaching using 1% sodium chlorite for 1 h at 70 °C. This step was repeated once again under the same conditions to obtain efficiently decoloured fiber material. After thorough bleaching, the material was treated with 5% solution of KOH at 25 °C for 15 h followed by centrifugation at 10 000 rpm for 20 min at 4 °C. The pellet obtained was redispersed in water and subjected to acid hydrolysis using 1% sulphuric acid solution for 1 h at 80 °C and centrifuged under similar conditions as briefed above. The pellet generated was redispersed in deionised water and then subjected to high intensity ultrasonication at 400 W, 800 W and 1000 W for 30 min in ice water bath. The obtained purified CNFs were freeze dried and stored at 4 °C.[60] Various structures of banana based CNF are given in Fig. 2.

3.3 Wood pulp based CNF

Wood based CNF are advantageous over other plant based CNFs due to its high cellulose purity, strong and ductile networks formed *via* CNF–CNF bonding of fibrils,[65] high intrinsic physical properties,[66] zero axial thermal expansion,[67] and high biodegradability.[68] CNFs have been isolated from both soft and hard wood pulps. The structure of hardwoods is more complex and heterogeneous than softwoods. Hardwood have specialized vessel elements involved in transport functions and shorter fiber cells when compared to softwood.[69] Due to the complex nature of the wood pulp, scientists have mostly used mechanical process for isolation of CNF from hard and softwood pulps. Stelte and Sanadi[70] isolated CNF from commercial hardwood

and softwood powders through mechanical fibrillation process by subjecting the wood pulps to initial refining followed by high-pressure homogenization. Isolation of CNF from hardwood pulps are difficult than softwood pulps and can affect the CNF isolation process through pressure fluctuations and clogging of the homogenizer. The number of passes used for homogenization process was kept high in hardwood pulp when compared to softwood pulp. The number of passes in hard wood pulp can be reduced by using appropriate sieves. Abe et al.[71] isolated CNF from radiata pine wood powder using simple mechanical process. The wood pulps subjected to grinding treatment in an undried state yielded CNF having uniform width of approximately 15 nm. Ultrafine polysaccharide nanofibers (5–10 nm) were prepared by treating microcrystalline α-cellulose obtained from wood pulp (Biofloc 92 MV) with TEMPO/NaBr/NaClO method.[72] These nanofibers were used to synthesis thin-film nanofibrous composite (TFNC) membranes useful for water purification. Yano et al.[73] successfully isolated CNF from soft wood (kraft pulp) and made composite films having high optical transparency using it as reinforcements in acrylic resins. The kraft pulp was bead milled followed by acetylation to reduce the hydrogen bond found between the natural CNF in pulp. Such acetylation process with decreased moisture content showed increased hydrophobicity. Terenzi et al.[74] used a combination of process to obtain CNF from softwood sulphite pulp. The softwood pulp was subjected to enzymatic degradation, mechanical beating and disintegration by passing through microfluidizer to obtain CNF of 6.6 nm diameter and more than 700 nm length. Galland et al.[75] prepared hollo CNF of high molar mass and small diameter that can be used for preparing high strength nanopapers. Aspen wood chips were chemically treated using chelating agents like diethylene triaminepentaacetic acid (DTPA)/sodium sulfite solution to remove metal ions.[76] Such treatment increase peracetic acid bleaching and cellulose stability.[77,78] Further the pulp was subjected to alkali treatment and mechanical disintegration using microfluidizer. The process yielded unique core shell CNF nanofibers having 0.2 nm "shell" made of hemicellulose (up to 24%) and 3.6 nm "core" CNFs. These CNF also had appreciable optical transparency and mechanical strength that can contribute to the strength of the CNF based nanopapers. Chen et al.[61] separated and individualized CNF (5–20 nm dia) from delignified hemicelluloses free poplar wood using combined chemical and high-intensity ultrasonication process (1000 W). The degradation temperature of these CNF increased from 210 °C to 335 °C when compared to that of original wood fibers. Needle-leaf bleached kraft pulp was mechanically disintegrated using a bead mill to get CNF.[11] The obtained CNF was further surface modified using alkenyl succinic anhydride for its efficient use as reinforcement matrix in high-density polyethylene. Zhao et al.[79] used high shear homogenisation process to extract CNF from dry softwood pulp. The extracted CNF were having a diameter ranging from 16–28 nm; however the thermal stability was low. Never-dried wood pulp was enzymatically treated using commercial enzyme (Novozym® 476) followed by subjecting to high shear forces through microfluidizer to obtain CNF of 0.8–2.0 μm

length and 10 nm dia.[80] Similar treatments combining enzyme and microfluidizer were carried out to extract CNF from never-dried spruce sulfite pulp.[59] The size of these nanofibers ranged from 2.5 to 10.5 nm.

3.4 Rice based CNF

Different parts of rice (*Oryza sativa* sp.) namely straw, stalk, husk *etc.* are an abundant source of CNF. The predominant component present in the cell wall of rice straw is cellulose. The cellulose chains are hydrogen bonded with each other resulting in formation of cellulose microfibrils with high tensile strength and crystallinity. Such cellulose microfibrils are further bonded with a gel matrix composed of hemi cellulose, lignin and other carbohydrates. This high crystallinity of the cellulose and the complex lignin and hemicellulose structure makes extraction of CNFs difficult from rice straw.[81–83] Different techniques have been employed for isolation of CNF from rice. Although there are several physical treatments available for extraction of cellulose from rice based plant materials like crushing and grinding, steam explosion has been reported to be effective for preparing the rice straw for chemical and enzymatic interaction.[84] Among various methods reported, steam explosion (hydrothermal treatment) in combination with acid-alkaline exposure has been reported to be efficient method for CNF production since the extracted CNF has high thermal stability (280 °C), excellent crystallinity, mechanical stability and remarkable purity. The exposure of rice straw to steam explosion leads to the breakdown of biopolymer and results in defibrillation of cellulose.[85,86] These highly stable CNF's have been considered as potential reinforcing material for composite preparation. Such CNFs have wide applications as drug delivery vehicles, reinforcing agents, soft tissue replacements and food packaging material.[5,86] The husk of the rice is also a potential source of CNF. Besides their thermal stability and crystallinity, the CNFs extracted from rice husk exhibited outstanding fluorescence emission capacity with remarkable quantum yield. Compared to other plant sources of CNFs, the intrinsic fluorescence ability is unique to rice husk.[87] The extracted CNF exhibited blue fluorescence when illuminated under UV light. The mechanical treatment and acid hydrolysis of rice husk led to the generation of CNF with trace amounts of lignin content having syringyl group and phenylcoumarone group which are responsible for its fluorescence property. Kalita *et al.*[87] reported the production of highly fluorescent CNFs from two rice varieties (*Oryza sativa* L. ssp. indica) namely Ahu and Boro. They followed hydrothermal treatment, acid-alkali exposure and mechanical disintegration to achieve CNFs of 35 nm. After washing and drying, the risk husk was made into fine particles by milling. The material was pre-treatment by soaking it with 2% NaOH for 14 h followed by autoclaving at 210 ± 5 °C and 20 lb pressure for 8 h with subsequent washing to remove the NaOH. In order to bleach the autoclaved fibers, the material was exposed to a mixture of NaOH–acetic acid and sodium hypochlorite solution. The bleaching of the fibers was followed by distilled water washing and drying. Further, the fibers were exposed to acid treatment by sonicating them in 10% solution of HCl at
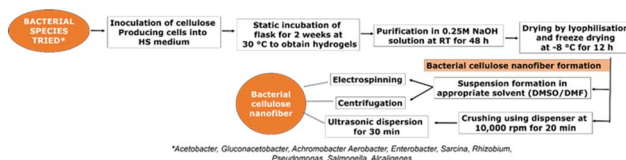

Fig. 3 Various steps involved in the extraction of bacterial CNFs.

40 °C for 2 h. This was followed by multiple steps of thorough washing and drying to obtain highly homogenous and fluorescent CNFs.[87] Similarly, Nasrabadi *et al.*[88] reported the extraction of CNF from rice straw using a chemo-mechanical method. The extracted CNF had a diameter of 70–90 nm with high crystallinity. The extraction protocol included alkali pre-treatment of the rice straw using NaOH (17.5 wt%) for 2 h followed by acid hydrolysis using diluted HCl at 80 ± 5 °C up to 3 h. The acid treated fibers were further subjected to alkali treatment (2 wt% NaOH at 80 ± 5 °C, 2 h) and air dried. After bleaching the treated fibers using sodium chlorite solution for 1 h, they were defibrillated by sonication (20 KHz) for 30 min in ice water bath. Chen *et al.*[61] demonstrated the extraction of CNF by hydrothermal treatment which included grinding and exposure to steam at 160 °C at 2 bar. Finally, the filtered and washed fiber residues were oven dried at 60 °C for 16 h to yield CNF.

3.5 Other sources of CNF

In addition to the above mentioned sources, CNF have been extracted from other agro and plant based sources like corn straw,[180] wheat straw, soy hulls, rhizomes, seagrass *etc.* Alemdar and Sain[89] reported the isolation of CNF from wheat straw and soy hulls using chemical–mechanical treatments. The CNF isolated from wheat straw had a diameter of 10–80 nm while soy hull based CNF had a diameter of 20–120 nm. This approach enhanced the cellulose extraction to 84%. The extracted CNF exhibited remarkable thermal stability (290 °C) and crystallinity. The extraction protocol briefly consisted of alkali treatment of the raw materials (NaOH 17.5% w/w for 2 h) followed by acid hydrolysis for 2 h (1 M HCl at 80 ± 5 °C). The treated material was once again subjected to similar alkali treatment and was filtered using vaccum before drying at RT. The dried fibers were then exposed to cryocrushing, defibrillation (2000 rpm) and high pressure homogenisation (>300 bar).[89] Ligno-cellulosic materials such as achira rhizomes also serves as an excellent source of CNF due to their specific properties and high cellulose content present in them. The isolated CNF are commonly used as fillers for films with high biodegradability.[50] Andrade-Mahecha *et al.*[50] demonstrated the development of various methods for the extraction of CNFs from the starch extract fibrous material of achira rhizomes. They introduced modifications in the bleaching process, acid hydrolysis and mechanical treatment steps to achieve high quality nanofibers. The achira rhizomes were dried, ground and sieved followed by alkaline treatment using 5% KOH for 13 h at RT. The content was chelated using 0.025% EDTA for 1 h at 70 °C and exposed to a couple of bleaching processes. The bleached materials were

subjected to second KOH treatment for 13 h at RT. Well dispersed CNF suspension was obtained after acid hydrolysis (1% HCl at 80 °C for 1–2 h) and high pressure homogenisation.[50] The CNF extracted were in the size range of 13.8–37.2 nm with high crystallinity (69.8%). They also had high negative surface charge which is an important factor in deciding the stable dispersion of CNF in suspension.[90] Effective isolation of CNF has been reported from sea grasses that can be utilised for the development of various bionanocomposites. The isolation of CNF from *Posidonia oceanica* leaves and balls have been reported using simple TEMPO mediated oxidation method followed by mechanical disintegration. Compared to other chemical treatments, TEMPO mediated oxidation is reported to have higher CNF yield (>90%).[91] They attempted to demonstrate the relation between concentration of the oxidant used and different grades of CNFs obtained based on it. In addition, they also combined the advantages of mechanical treatment, ultrafine grinding along with TEMPO oxidation to achieve highly disintegrated CNFs. Briefly, the raw materials were dissolved in sodium bromide and TEMPO containing distilled water and the reaction was initiated by the introduction of NaClO into the suspension under constant stirring at RT. pH was maintained at 10.0 using NaOH. Ethanol was added to the suspension finally and the cellulose fibers produced were filtered and washed multiple times using distilled water. The obtained fibers were nanofibrillated after dispersing it in water using a high intensity ultrafine grinder at 2500 rpm.[91]

Chen *et al.*[25] demonstrated efficient CNF extraction from different sources such as newspaper, corn straw, bamboo, wood and rice straw using three different mechanical nanofibrillation techniques namely, blender, ultrasonicator and high pressure homogenizer. All raw materials were collected, sieved and air-dried. The materials were dewaxed for 6 h in mixture of benzene : ethanol solution (2 : 1). Later, the samples were acidified multiple times for 1 h at 75 °C using sodium chlorite solution and subsequently treated with 3% KOH for 1 h at 90 °C. Another cycle of acidification and alkaline treatment using KOH was carried out under the same conditions as mentioned above to produce highly purified cellulose pulp. For nanofibrillation of cellulose pulp using a blender, the pulp was made into suspension using distilled water and agitated for 20 min using a domestic blender. In case of ultra-sonication, the cellulose suspension was subjected to high intensity ultra-sonication for 20 min at 1200 W. Similarly, during high pressure homogenisation, the purified cellulose suspension was first fibrillated using high intensity ultrasonicator for 5 min at 1000 W. These samples were then exposed to high pressure homogeniser for 20 min.[25] The benefits of adopting a combination of chemical and mechanical treatment for the extraction of CNFs were reported by Sehaqui *et al.*[92] In their study, cationic CNFs were isolated from pulp residues. The pulp material was beaten mechanically and allowed to react with aqueous NaOH to result in suspension. Under constant stirring, glycidyltrimethyl ammonium chloride was introduced to the suspension at 65 °C and the reaction was allowed to proceed for 8 h. It was then treated with HCl followed by filtration and washing. Finally, the resulting material was dispersed in water and agitated well for

10 min. This was later taken for high shear homogenisation for the disintegration of cellulose fibers at 1200 bar pressure.[92] Similarly, Theng *et al.*[93] demonstrated the production of CNFs from *Eucalyptus* pulp and corn biomass using TEMPO mediated oxidation. The corn pulp production comprised of chopping the corn biomass and sieving those using 10 mm mesh sieve. They were then boiled in a digester for 15 min at 160 °C using distilled water in ratio of 6 : 1. The obtained pulp was washed well and filtered using a Sprout-Waldron refiner. The resulting pulp material was vaccum dried and stored at RT. The TEMPO oxidation carried out were similar to the protocol followed by Soni *et al.*[17] After TEMPO mediated oxidation, the resulting fibers were exposed to mechanical disintegration using high pressure homogeniser operating at 600 bar pressure for 5 min at 60–70 °C. This resulted in the formation of CNF gels.[93]

4. CNF derived from bacterial sources

4.1 Natural bacterial CNF synthesis

CNFs are produced by various bacterial species. Gram positive bacteria belonging to species such as *Gluconacetobacter hansenii* and Gram negative species such as *Agrobacterium*, *Achromobacter*, *Aerobacter*, *Enterobacter*, *Sarcina*, *Rhizobium*, *Pseudomonas*, *Salmonella* and *Alcaligenes* spare well known in this category. Lee *et al.*[94] has reviewed in detail the metabolic and genetic pathway involved in the production of bacterial CNFs. The review also shed light on various bioreactor systems being employed for the production of CNFs using bacteria.[94] Different forms of bacterial CNF structures such as spheres, gels, sheets, membranes, mats *etc.* can be produced by introducing simple modifications in the production strategy. Detailed descriptions of various steps involved in the synthesis of bacterial CNFs are given in Fig. 3.

Mohite *et al.*[95] reported the synthesis of bacterial cellulose from *Gluconacetobacter hansenii* NCIM 2529 using shaking culture method for various environmental applications. The organism was maintained in a medium containing yeast extract, mannitol, peptone and agar at a pH 5.5. For the production of cellulose, the organism was cultured in Hestrin and Schramn (HS) medium which was composed of citric acid monohydrate, glucose, peptone, yeast extract and $Na_2HPO_4 \cdot 12H_2O$ at pH 5.0. Their production protocol involved inoculation of the organism in the HS medium and culturing them for 120 h at 30 °C at a speed of 120 rpm. After incubation, bacterial cellulose containing broth was obtained and filtered, resulting in bacterial cellulose beads which were further purified using 0.1 N NaOH for 30 min at 90 °C. This is followed by filtration and thorough washing using distilled water. The resulting cellulose was finally dried in the oven for 8 h at 50 °C. The obtained CNF was reported to be white, translucent and oval shaped spheres containing micro and nano sized fine fibers. The resulting fibers exhibited high elasticity, surface area, wettability and strength.[95] Similar method was adopted by Fang *et al.*[96] for the production of CNFs using *G. xylinus* that were genetically modified using *Agrobacterium* sp. ATCC31749. The genetically transformed organism was cultured in HS medium at 30 °C for 2.5 days with 0.1% additional cellulose. The bacterial cells

Table 3 Shows various CNF based composites, their sources, preparation methods and properties

Composite composition	Source of CNF	Method of preparation	Properties of the material developed	Ref.
Modified cellulose composites				
Anthocyanins from *B. oleracea* L, cellulose acetate	Commercially obtained	Electrospinning	Size; 228 ± 118 nm, high stability at −50 °C, 100 °C and RT, pH responsiveness (visual colour differences) at pH 7–8	157
Cellulose acetate, polyamides b, poly vinyl acetate, poly acrylonitrile	Commercially obtained	Electrospinning	Moderate filtration efficiency	158
Cellulose acetate, poly hydroxyl butyrate	Commercially obtained	Electrospinning	Size; 80–680 nm, high porosity (81 ± 2.1% to 85 ± 2.6%), high tensile strength (5.05 ± 0.52 MPa), high yield strength (4.6 ± 0.82 MPa), biodegradability, high osteoblast compatibility	159
Cellulose acetate, poly vinyl alcohol	Commercially obtained	Electrospinning	Size; 251–368 nm, high tensile strength (9.8 ± 1 MPa), high entrapment efficiency (64.96 ± 3.14%), high permeation of capsaicin through skin model (shed snake skin), moderate human foreskin fibroblast (NHF) compatibility	135
Cellulose acetate, polyacrylonitrile	Commercially obtained	Electrospinning	Excellent flexibility, enhanced mechanical strength	160
Cellulose acetate, chitosan, oxolane-2,5-dione, furan-2,5-dione, furan-2,5-dione	Commercially obtained	Electrospinning	High metal adsorption capacity (221 $\mu g\ L^{-1}$-lead)	161
Cellulose acetate, carbon	Commercially obtained	Electrospinning	Size; 0.5–1.5 μm, hydrophilicity, highly amorphous nature, high specific surface area (25 $m^2\ g^{-1}$), moderate electrical conductivity (6.3 S cm^{-1}), high specific capacitance (20.9 F g^{-1})	162
Cellulose acetate, (AgNP), (RuNP)	Commercially obtained	Electrospinning, wet reduction method	Size; 325 ± 2.0 nm, high catalytic activity, high selectivity in oxidation of benzyl alcohol (100%), high stability and reusability	108
Carboxy-methyl cellulose, cellulose acetate, silver (AgNP)	Commercially obtained	Electrospinning	Size; 370 ± 174 nm and 410 ± 195 nm, high stability, selectivity, biocompatibility, high electrical conductivity, lower detection limit (1.64 μm)	106
1,4-Dihydroxyanthraquinone (1,4-DHAQ), cellulose acetate	Commercially obtained	Electrospinning	High selectivity and sensitivity in detecting Cu^{2+}, high reusability, favourable detection range of Cu^{2+} (2.5×10^{-9} to 3.75×10^{-8} M), lower detection range of Cr^{3+} (2.5×10^{-9} to 2.5×10^{-8} M)	163
Oxolane-2,5-dione, cellulose acetate	Commercially obtained	Electrospinning	Size; 384.86 nm, high surface area (13.68 $m^2\ g^{-1}$), reusability, high adsorption capacities (1.0–2.91 mmol g^{-1}), cost effective	127
Hydroxyapatite (HAp), cellulose triacetate	Commercially obtained	Electrospinning	Size; 346 to 815 nm, high protein (BSA) adsorption capacity (18.39 mg g^{-1})	109
Poly caprolactone, cellulose acetate	Commercially obtained	Co-electrospinning	Size; 700–850 nm and 300–550 nm, high mechanical strength, enhanced wicking rate	164
Hydroxypropyl cellulose, PCL, sulfisoxazole–cyclodextrin complex	Commercially obtained	Electrospinning	Size; 90 ± 40 nm and 60 ± 25 nm, high conductivity, high viscosity	134
Modified cellulose, poly vinyl alcohol	Commercially obtained	Electrospinning	Size; 117–500 nm, enhanced glass transition temperature (T_g = 15.6 °C), low crystallinity (52.10%), high mechanical properties	132
Nanofibrillated cellulose, hydroxyl ethyl cellulose	Never-dried softwood sulphite pulp fibers	Enymatic treatment, mechanical disintegration	Size; 15 nm, enhanced ductility, moderate stress yield of 8 MPa, enhanced strength of 80–93 MPa, increased stiffness, low Young's modulus (0.8–1.3 GPa), low storage modulus (<60 MPa)	65

Table 3 (Contd.)

Composite composition	Source of CNF	Method of preparation	Properties of the material developed	Ref.
Bacterial cellulose composites				
Bacterial cellulose, polyethyleneimine	Commercially obtained	Flush coating and cross linking	Large surface area, high porosity, high regeneration capacity, high adsorption of Cu^{2+} (90.1 mg g^{-1}) and Pb^{2+} (130 mg g^{-1})	105
Bacterial cellulose, poly vinyl alcohol	Commercially obtained	—	Enhanced pore structure, enhanced heat resistance (209 °C), high Young's modulus (1.0 MPa), high tensile strength (4.73 MPa), enhanced compression modulus (0.87 MPa)	133
Bacterial cellulose membranes, acrylate polymers	*Gluconacetobacter sacchari*	Conventional static culture technique, *in situ* atom transfer radical polymerization	High hydrophobicity (contact angle 134°), enhanced thermal stability (241–275 °C), high flexibility	165
Carbon nanofiber, bacterial cellulose	Bacterial cellulose polypyrrole	Mechanical treatment, carbonization reaction	Excellent electrochemical performance, high reversible specific capacity (240 mA h g^{-1}), enhanced rate performance (146.5 mA h g^{-1}), high cycling stability (148.8 mA h g^{-1} over 400 cycles)	104
Bacterial cellulose, silver (AgNP)	*Acetobacter xylinum* NUST5.2	Conventional static culture	Size; 30 nm, high anti-bacterial (*E. coli* and *S. aureus*) activity (inhibition zone 32 mm), favourable crystallinity, high thermal stability (250–290 °C)	107
Natural cellulose composites				
Natural cellulose, high density poly ethylene	Needle leaf bleached kraft pulp (NBKP)	Mechanical disintegration, injection molding	High tensile strength (43.4 MPa), high Young's modulus (1.97 GPa), excellent mechanical strength	11
Natural cellulose, poly lactic acid	Wheat straw	Chemi–mechanical treatment, high speed homogenisation	High crystallinity at high pressure condition, high viscosity	166
Cellulose, starch	Rice straw	Chemo-mechanical method, film casting, salt leaching, freeze drying	Size; 40–90 nm, high chondrocyte compatibility, biodegradability	88
Polyurethane, cellulose	Rachis of date palm tree (*Phoenix dactylifera*)	Mechanical treatment, high intensity homogenising, solvent exchange method	Size; 29 ± 9 nm, high degree of crystallinity (40.0%), moderate thermal stability (51 °C), high tensile strength	167
Cellulose, hemicellulose	Spruce sulfite pulp (commercially obtained)	Enzyme treatment, mechanical disintegration, filtration, drying	Size; 190 nm, enhanced storage modulus (>20 GPa), high thermal stability (300 °C), favourable Young's modulus (33 MPa), moderate tensile strength (3 MPa), high work of fracture (560 kJ m^{-3})	168
Cellulose, starch	Kenafbast fibers (*Hibiscus cannabinus*)	Solution casting	High tensile strength (38.0 ± 3 MPa), high Young's modulus (141.0 ± 35 MPa), moderate elongation at break (27 ± 4%), biodegradability	169

Table 3 (Contd.)

Composite composition	Source of CNF	Method of preparation	Properties of the material developed	Ref.
Cellulose, polyester resin	Softwood (*Pinus* sp.) and hardwood (*Eucalyptus* sp.)	Mechanical treatment	Size; 70–90 nm, high thermal stability (374 °C), moderate crystallinity index (72.8%)	170
Unsaturated polyester, cellulose	Never-dried wood pulp (Nordic Paper, Sweden)	Mechanical treatment, template-based processing approach	Size; 100–200 μm, enhanced storage modulus, thermal stability up to 60 °C, high glass transition temperature (T_g = 78 °C), high moisture sensitivity	80
Cellulose, amylopectin	Spruce sulphite pulp (Nordic pulp and Paper, Sweden)	Enzyme degradation, mechanical treatment, disintegration using microfluidizer	Size; 68 nm, 361 nm, 186 nm, moderate Young's modulus (13.6 GPa), high yield strength (117 MPa), enhanced strength (221 MPa)	171
Cellulose, polyacrylamide	Fibrous cellulose powder-CF11 (commercially obtained)	Acid hydrolysis	High compressive stress (4.43 ± 0.06 kPa), hydrophilicity, high mechanical strength, favourable thermal stability (286–289 °C)	160
Cellulose, polyaniline, carbon nanotubes	Bamboo powders from moso bamboo	Chemical treatment, *in situ* chemical polymerization	Size; 10 to 30 nm, flexible, foldable, high specific capacitance (249.7 F g^{-1})	172
Cellulose, multi-walled carbon nanotubes, polyaniline	Bamboo powder from moso bamboo	Chemical treatment, solvent extraction, *in situ* polymerization	Size; 10–30 nm, high specific capacitance of 791.13 F g^{-1}, high porosity, superior cycling stability, high redox reversibility	143
Cellulose, carbon nanotubes, TiO$_2$ nanotubes	Bamboo cellulose tissues	Mechanical treatment	Size; 10–30 nm, high porosity, high mechanical strength, superior discharge capacity (62.5 mF cm^{-2})	103
Cellulose, cadmium sulphate (CdS)	Natural cotton	Electrospinning, chemical bath deposition	Size; 100 nm, amorphous nature, high photocatalytic activity	56
Titanium dioxide (TiO$_2$), cellulose, gold (Au), silver (Ag)	*Eucalyptus* pulp (USDA Forest Service-Forest products Laboratory (Madison, WI)	TEMPO mediated oxidation, mechanical treatment	Size; 4–20 nm, superior mechanical properties, reusability, high Young's modulus (17.7 ± 6.6 MPa), high tensile strength (70.7 ± 12.1 MPa), enhanced photocatalytic activity	126
Cellulose, quaternary ammonium	Softwood kraft pulp	Mechanical treatment	Size; 10–40 nm, high porosity, high reusability (84.9% after 4 cycles of adsorption)	128
Others				
Polyethylene-*b*-poly(ethylene glycol), cellulose	Commercially obtained	Spray drying, surface adsorption, extrusion	High tensile modulus	173
Cellulose, poly vinyl alcohol	Micro crystalline cellulose (commercially obtained)	Acid hydrolysis	High water resistance, enhanced thermal stability (100 °C), size; 10–65 nm	174
Cellulose, poly lactic acid	Commercially obtained	Solvent casting	Size; 28 ± 10 nm, high thermal stability (156.3 °C), high tensile strength (33.1 MPa), high elastic modulus (188.9%)	175

Table 3 (Contd.)

Composite composition	Source of CNF	Method of preparation	Properties of the material developed	Ref.
Cellulose, poly(lactic acid)	Nano Novin polymer co. (Iran)	Solution casting method	Size; 21 nm, high crystallinity (72%), high thermal stability (260.5 °C), high maximum degradation temperature (290.8 °C)	176
Cellulose, starch, poly vinyl alcohol	Microcrystalline cellulose (commercially obtained)	Acid treatment, solution casting	Size; 20–35 nm, excellent mechanical properties, high strength (19.5 MPa), high stiffness (1199 MPa)	141
Polyethylene oxide, cellulose nanocrystal	Microcrystalline cellulose (commercially obtained)	Acid hydrolysis, high pressure homogenisation, electrospinning	Size; 149 ± 49 nm, high Young's modulus (37.9 ± 0.6 MPa), high glass transition temperature, enhanced elongation at break (200 ± 17%)	177
Cellulose, copper (Cu^{2+})	Cellulose sludge (commercially obtained)	Mechanical treatment, TEMPO mediated oxidation	Size; 15–40 nm, enhanced wettability, hydrophilicity, high Cu^{2+} adsorption capacity (75 mg g^{-1})	124

obtained by high speed centrifugation was washed thoroughly using concentrated glycerol and was further maintained at 80 °C. This method produced CNFs that were highly crystalline in nature.[96] Gonçalves et al.[97] reported the production of bacterial CNF using static culture technique. In this work, bacterial CNF were produced from G. xylinus ATCC 53582. The organism was cultured for 30 days in HS medium. The bacterial cellulose sheets produced were purified using 0.1 N NaOH. After thorough washing using distilled water, the cellulose sheets were cut in to appropriate sizes and dried at 50 °C for 8 h. Biocompatible, highly porous, ultrafine CNFs were produced using this technique.[98] Similarly, Quero et al.[98] reported the production of bacterial CNF gel using static culture. In their study, G. xylinum 13693 was cultured in the HS medium for 14 days at 27 °C during which highly fibrous cellulose gel was formed. The gel was then squeezed in sterile environment to produce suspension of cells. To the main culture medium, the prepared cell suspension was inoculated and maintained for 14 days. The formed bacterial cellulose fibrous structure was then purified using NaOH and distilled water. In order to remove the water content completely from the produced cellulose network, their group adopted the method of hot pressing at 120 °C for 4 min.[98] Apart from cellulose gels, mats of bacterial CNFs were also produced using static culturing technique. For instance, Olsson et al.[99] developed a protocol for the synthesis of bacterial cellulose mats containing microfibers and nanofibers using G. xylinus (ATCC 23767). The procedure comprised of initial inoculation of the bacteria in 2 mL culture medium to produce initial suspension containing cellulose. After 3 days, the cellulose suspension obtained was used as the inoculum for second cycle of culturing in higher volume of growth medium. This cycle was repeated with higher volumes of culture medium until final inoculation in 20 L of medium. Pure cellulose mat generation was observed after 2 days of final inoculation in the medium while a fully grown cellulose mat was harvested by the 7th day. The obtained mats were further purified by boiling in NaOH solution and washing with distilled water. The water content was completely removed by compressing the mats into thin sheets after which it was stirred continuously at 60 °C for 72 h in aqueous H_2SO_4. This step completely dissolved the cellulose in the solution. To obtain CNF suspension, the acid dissolved cellulose suspension was subjected to 3 cycles of centrifugation under isothermal conditions at 24 000 rpm. The final cellulose pellet obtained was redispersed in distilled water or appropriate solvent using high shear mixer for 5 min.[99]

In another study, Lee et al.[100] reported the extraction of bacterial CNFs from coconut gel using Acetobacter sp. The synthesis protocol comprised thorough blending of the gel contents using a high speed blender after they were washed carefully using deionised water. The contents were then subjected to high speed homogenisation at 20 000 rpm followed by centrifugation at 14 000g. The purification of bacterial CNFs were then carried out by boiling the water dispersed cellulose content in NaOH for 20 min. A final centrifugation step was carried out to obtain highly homogenous bacterial CNFs. This procedure resulted in the formation of nanofibers of 50 nm size.[100] Acetobacter xylinum derived bacterial CNF hydrogel was

produced by Bäckdahl *et al.*[101] for the synthesis of highly porous tissue engineering scaffold. In his work, sub species of *A. xylinum* sucrofermentas BPR2001 was used as the source of CNF. Corn steep liquid media was used as both culture and production medium.

5. CNF composites

CNF obtained from both plant and bacterial sources has been combined with various metals and non-metals such as carbon, nanoparticles, fluorophore, polymers *etc.* to prepare composites for numerous applications (Table 3). The high mechanical strength and integrity of the CNF provide remarkable properties to the composites. Such composites find immense applications in water purification, food packaging, supercapacitor preparation, tissue engineering *etc.* A detailed view on various CNF based nanocomposites has been given in Table 2. Association of CNF with carbon nanotubes are well known for their application as supercapacitor electrodes. CNF/carbon nanotubes were blended with polyaniline (PANI),[102] TiO$_2$ nanotubes[103] and nitrogen[104] for achieving high mechanical strength, flexibility, excellent cyclability, high discharge capacity and remarkable specific resistance. Besides flexible electronics, they were also used in wearable textiles.[103] CNF for such applications were obtained from plant sources such as bamboo tissues using mechanical grinding at 1500 rpm (ref. 82) and from bacterial cellulose.[104] Fig. 4 and 5 shows SEM images of composite CNF obtained using bacterial CNF and plant based CNF.
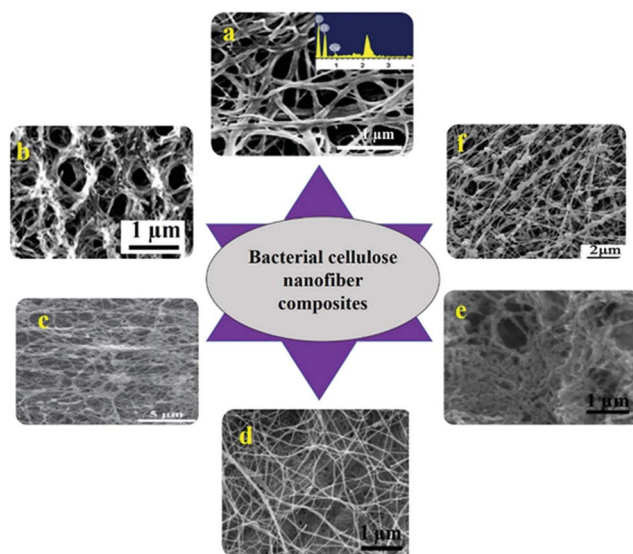
The CNF composites has been prepared using *in situ* chemical polymerization technique, oxidative polymerization, freeze drying with *in situ* polymerization[102] and hydrothermal technique. Combination of plant derived CNF/carbon nanotubes/PANI and CNF/carbon nanotubes/TiO$_2$ resulted in fibers with diameters of 10–30 nm while bacterial cellulose/carbon nanotubes resulted in fibers with 30–60 nm diameter. Association of CNF with fluorophores has been reported by Wang *et al.*[105] where CNF/1,4-dihydroxyanthraquinone (1,4-DHAQ) composite has been employed for effectively detecting Cu^{2+} and Cr^{3+} in contaminated water.[105]

5.1 Nanoparticle based CNF-composites

CNF has been combined with various nanoparticles such as CdS nanoparticles, silver nanoparticles, hydroxyapatite (HAp) nanoparticles, gold nanoparticles *etc.* for catalytic applications. CNF with silver nanoparticles has been used as biosensors for the detection of catechol,[106] preparation of antibacterial fibers[107] and catalytic materials.[108] They have been prepared by wet reduction method using NaBH$_4$, *in situ* chemical reduction and simple chemical binding techniques. In anti-bacterial fiber preparation, bacterial cellulose was doped with silver nanoparticle to obtain fibers of 1.5 nm that showed high resistance against the growth of *Escherichia coli* (*E. coli*) and *Staphylococcus aureus* (*S. aureus*).[107] CNF from natural cotton were reported to be functionalized with CdS nanoparticles by chemical bath deposition method for photocatalysis of organic pollutants.[56] Briefly, the synthesis method consisted of soaking the electrospun CNF in Cd(NO$_3$)$_2$ ethanol solution followed by Na$_2$S solution each for 3 min. It was further dried at 60 °C to obtain CdS/CNF of 100 nm diameter. Hap nanoparticles (HAp/CNF)

Fig. 4 SEM images of various nanocomposites made of bacterial cellulose (a) commercial BC/polyethylenimine nanofibers adsorbed with cu^{2+} (ref. 105) (b) cross sectional image of PMMA grafted on BC nanofibers obtained from *Gluconacetobacter sacchari*[106] (c) *Acetobacter xylinum* produced CNF coated with calcium phosphate[107] (d) cytochrome C coated CNF processed from *A. xylinum* (ATCC 10245)[108] (e) commercially obtained bacterial cellulose template synthesized carbon nanofiber[104] (f) lecithin immobilized BC produced by *A. xylinum* X-2 (ref. 109) (all images were reproduced with permission).
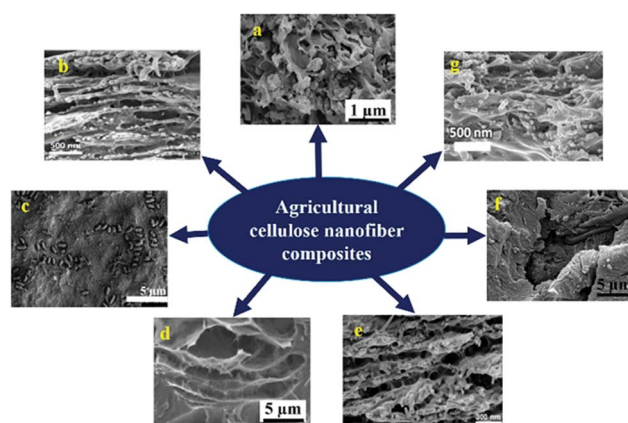

Fig. 5 SEM images of various nanocomposites delivered using plant CNF (a) softwood derived CNF and PLA blend[110] (b) core shell nanofiber obtained by blending spruce sulphite pulp derived CNF and amylopectin (c) soft wood sulphite pulp derived nanofibrillated cellulose–HEC composite[119] (d) palm tree ranchis processed CNF and polyurethane blend[172] (e) Core–shell image of hemicellulose and spruce sulphite pulp extracted CNF blend[171] (f) *Pinus* sp. and *Eucalyptus* sp. extracted cellulose nanofiber–polyester blend[173] (g) composite obtained from 1-butyl-3-methylimidazolium chloride (BMIMCl) and canola straw derived CNF[174] (all images were reproduced with permission).

has been reported to be used for protein purification applications. Fig. 6 shows SEM images of various modified CNF based composites. Tian et al.[109] demonstrated the synthesis of HAp nanoparticles functionalized cellulose triacetate nanofibers for the purification of bovine serum albumin. The electrospun hybrid nanocomposite was reported to have a core (cellulose triacetate)–shell (HAp nanoparticles) structure with diameter of 346–816 nm and they exhibited high adsorption of BSA.[109]

5.2 Synthesis of modified bacterial CNF

Bacterial CNFs are well known for its characteristic properties such as high crystallinity, tensile strength, high surface area, nano sized network structure.[110] In order to further enhance their efficiency for various applications, CNFs surface has been modified using numerous functional groups. Various modification carried out with bacterial CNFs are provided in the following section.

5.2.1 Polymer functionalization

5.2.1.1 Amine. The surface modification of bacterial cellulose using amine groups were reported to enhance the surface area and adsorption capacity of the resulting modified CNF. Wang et al.[110] demonstrated the use of polyethyleneimine (PEI) surface functionalised bacterial CNF in the adsorption of heavy metal ions and organic dyes from contaminated water sources. They reported simple flush coating method followed by heat treatment for the functionalization of PEI on bacterial CNFs. The method comprised thorough washing of bacterial CNF membrane with sodium hydroxide (0.1 M) at 100 °C followed by flushing the membrane with prepared PEI solution (10.0 g) under vacuum and later heating at 70 °C for 30 min. The unbound PEI was removed from the surface of cellulose membrane by washing with deionised water followed by vacuum drying.

5.2.1.2 Aniline. To develop CNF based materials for various electronic applications, bacterial CNFs were functionalized using polyaniline that imparts the material excellent conductive properties. Such nanocomposite materials exhibited outstanding potential as supercapacitors. One such work has been reported by Wang et al.[111] where pristine bacterial CNF obtained by shake culture method were surface polymerised with polyaniline. The preparation protocol included dispersion of synthesised bacterial CNF in solvent having combination of DMF and distilled water at 25 °C. 1 mL of prepared aniline monomer was mixed vigorously with the cellulose suspension followed by cooling. This ensured self-assembly of aniline on the surface of the nanofibers. This was followed by the drop wise mixing of ammonium peroxide sulphate and hydrochloric acid mixture into the nanofiber suspension under constant stirring. The reaction was left to proceed overnight and allowed to precipitate which was filtered and washed with combination of acetone, water and HCl. This was followed by drying overnight to achieve green coloured solid nanocomposite material, containing bacterial CNF of 30 nm size with flake morphology. The produced composite showed high electrical conductivity and enhanced surface area.[102] Similar protocol was adopted by Hu et al.[112] to develop bacterial CNF based flexible membranes

that were surface coated with polyaniline nanoparticles. The synthesised nanocomposite membrane exhibited high thermal stability, electrical conductivity and excellent mechanical properties. Such conductive bacterial cellulose nanocomposite materials find extensive applications in flexible electronics such as electrodes, displays and sensors.[112] Likewise, Lin et al.[113] demonstrated the formation of bacterial cellulose–polyaniline film with single conductive side to be used in supercapacitors. The synthesis protocol of the nanocomposite film included initial wetting of bacterial cellulose film and wrapping it around a mould such as cup. The prepared aniline–toluene solution was then poured inside the mould. For 10 min, the prepared mould was placed within the aniline monomer and ammonium persulphate solution which was maintained at 0 °C for 1 day that resulted in polyaniline formation at the bottom of the mould. Further the mould was removed from the APS solution. Using vacuum filtration, the aniline–toluene solution and remaining contents from within the mould were carefully removed. Mixture of acetone and distilled water was used to wash the inner region of the mould followed by vaccum drying for 1 day at 60 °C. Later, the bacterial cellulose–polyaniline composite film with single phase coating was carefully obtained from the bottom of the mould.[113]

5.2.1.3 Methacrylate. Bacterial CNFs have immense applications as templates and scaffolds that assist cell growth in tissue regeneration. In order to enhance their suitability as scaffolds, these nanofibers were blended with various polymers such as methacrylates including acrylic acids, 2-ethylhexyl acrylate, 2-hydroxyethyl methacrylate which has high

Fig. 6 Nanocomposites obtained using modified CNF (a) SEM image of cellulose acetate/1,4-DHAQ/CA nanofiber film (insert-high magnification image of cellulose acetate/1,4-DHAQ/CA nanofiber)[175] (b) cellulose acetetate–poly hydroxybutyrate blended nanofibers[180] (c) hydroxyapatite decorated cellulose triacetate nanofibers[179] (d) cellulose acetate nanofibers decorated with AgNP[178] (e) cellulose acetetate–PCL blended nanofibers after deacetylation[181] (f) polyamide layer coated on PAN/cellulose acetate nanofibers (inset (10 000×) image of polyamide surface)[182] (all images were obtained with permission).

biocompatibility with human cells and tissues. One such work has been reported by Hobzova et al.[114] where bacterial CNF–methacrylate composite hydrogels suitable for various tissue engineering applications were fabricated using simple polymerisation method. The CNFs were produced from *Acetobacter xylinum* sp. (sub species: sucrofermentas BPR2001) that was cultured in corn steep liquor medium and the synthesised nanofibers were maintained in swollen state. For the formation of composite hydrogels, the bacteria derived CNF sheets were immersed in the polymerisation mix at RT for 1 day in the presence of nitrogen gas under shaking condition. Later, discs were framed using silicone after carefully placing them on suitable glass substrate. A polypropylene plate was used to cover the sheets which were then closed in to a mould using appropriate clamps. This was followed by 20 min exposure to UV illumination for polymerisation. The compound thus formed was carefully removed from the mould and washed consecutively for 5 days using distilled water.[115] Apart from the biological applications, bacterial CNFs with methacrylates also find suitable applications in flexible electronics. Nogi and Yano[115] reported the fabrication of thermally stable and foldable flat panel display (FPD) by reinforcing acrylic resin with bacterial CNFs. In their work, *A. xylinum* FF-88 was used to produce CNFs under static conditions. The cellulose pellicle obtained after 10 days were purified by boiling in NaOH for 2 h with subsequent washing using water for 2 days. The final product obtained was compressed and immersed in ethanol : water solution (50%) and the content of ethanol was gradually increased until 100% to completely replace the solvent present in the cellulose pellicle. Further, the acrylic resin was impregnated with the CNF pellicle for 12 h under pressure of 0.09 MPa. Ultraviolet curing was finally carried out to cure the resin.[115] Similar method of solvent exchange was adopted by Kramer et al.[116] for the fabrication of nanocomposites where bacterial CNFs were photopolymerised with methacrylate and acrylate *via* methacrylate crosslinking. The CNFs were obtained by the static culture of *G. xylinus* (strain: DSM 14666) and the developed composite material exhibited collagen like properties such as high water absorption, elasticity, biocompatibility, nanofibrous structure and modifiable pore size.[116]

5.2.1.4 Poly vinyl alcohol (PVA). The combination of bacterial CNFs with PVA resulted in nanocomposites with extraordinary mechanical strength, thermal stability and toughness. Bacterial CNF–PVA hydrogel was synthesised by mixing PVA powder to the nanofibrillated suspension with distilled water and stirred for 2 h at 85 °C in the presence of a crosslinker, gluteraldehyde.[10] The obtained hydrogel was washed in deionised water and the surface water content of the hydrogel was removed by gentle tapping using a clean filter paper.[10] Likewise, Millon and Wan[117] optimised protocol for fabricating bacterial CNF–PVA composite hydrogel as replacement for cardiovascular tissues. In his protocol, *A. xylinum* was cultured to synthesise cellulose fibers in nano size ranges. CNF–PVA composite solution was prepared according to the method reported by L. Yang et al.[10] Further, this solution was transferred into appropriately designed moulds of aluminium and was placed in a refrigerator (0.1 °C min⁻¹) with multiple freeze and thaw cycles. His group

has also demonstrated the efficiency of bacterial CNF–PVA composite material developed for cartilage replacement following the same protocol.[117]

5.2.1.5 Poly ethylene oxide (PEO). Bacterial CNF in combination with PEO resulted in high quality nanocomposites useful for various applications. Brown et al.[118] fabricated stable and flexible thermoplastic material using bacterial CNF from *A. xylinum* and PEO. Briefly, *A. xylinum* 23769 was inoculated and maintained in autoclaved HS medium of pH 5.0. The bacterial culture was carried out under static conditions. The cellulose pellicle formed in the medium after one week of culturing was obtained carefully and processed. Later, this was inoculated into a separate HS medium containing PEO. After two days of static maintenance, the cellulose pellicle obtained was carefully removed. The leftover medium was cultured again under agitation of 500 rpm. After two days of culture, the strings of cellulose formed was collected carefully using a gauge and washed multiple times using deionised water.[118]

5.2.2 Metal functionalization

5.2.2.1 Palladium. Metal particles were doped on the surface of bacterial CNFs to enhance its properties for various applications. Such surface metal doping enhanced the mechanical features and stability of the resulting CNF structure. For instance, bacterial CNF doped palladium has been proved to be an efficient catalyst for Heck coupling by Zhou et al.[119,178] The preparation protocol involved hydrothermal reduction method where palladium particles were physically coated on the surface of bacterial CNFs. In detail, $PdCl_2$ was mixed well with the aqueous solution of bacterial CNFs followed by degassing the solution for 30 min and subsequent heating under the influence of N_2 at 140 °C for 5 h with constant stirring. Further, aqueous solution of potassium borohydride was added into the mixing solution and was allowed to mix at 80 °C for 3 h after which, it was subjected to multiple cycles of centrifugation and washing resulting in black coloured nanocomposite.[119]

5.2.2.2 Silver. Bacterial CNFs were also stable templates for the surface doping of silver particles. This enhances the stability, conductivity and mechanical performance of the resulting CNFs. This was explored by Ifuku et al.[120] where he and his co-workers synthesised bacterial CNFs using TEMPO mediated oxidation. The ion exchange reaction between the added Ag particles and the carboxylate groups on the fiber surface resulted in the attachment of Ag particles on the CNF. This was followed by metal reduction on to the surface as fine Ag nanoparticles. In detail, prepared bacterial cellulose fibrous sheets were immersed in aqueous solution of TEMPO and sodium bromide. To initiate oxidation reaction, sodium hypochlorite solution was introduced to the solution and the reaction was proceeded overnight at pH 10.5. Ethanol was later added to the solution as quencher to stop the reaction. The product was then dried overnight at 65 °C. In order to coat the bacterial CNF surface with Ag nanoparticles, prepared nanofiber was immersed in aqueous solution of $AgNO_3$ at RT in dark and the reaction was allowed to proceed overnight. The reduction reaction was performed at 100 °C for 1 h under RT.[120]

5.2.2.3 Gold. Bacterial CNFs are well known for their association with gold nanoparticles (AuNP). Chen et al.[121]

demonstrated the efficiency of bacterial CNF as templates for the growth of AuNP on their surface. The developed nanocomposite was reported to have high catalytic activity towards the reduction of 4-nitrophenol and was proposed to have wide range of industrial applications. In this work, CNFs were obtained from *G. xylinum* and were made in to a suspension using deionised water using a supercollider. Chen *et al.* introduced a two-step method for preparing the CNF substrate for AuNP production. Initially, cellulose activation was done by dispersing it in NaOH solution for 7 h. Further, under constant stirring conditions, acrylonitrile was gradually added at RT and the reaction was allowed to proceed for 12 h. Later, cycles of centrifugation and washing were carried out to obtain cyano ethyl surface modified bacterial cellulose. This was followed by dispersion of surface functionalised cellulose in a solution of $NH_4OH/NaOH : HCl$ for 10 h at 50 °C. This resulted in amidoxime surface functionalised CNFs (AOBC). The obtained cellulose fibers were purified from other components by multiple rounds of centrifugation-washing, followed by freeze-drying. The freeze-dried AOBC was dispersed in deionised water containing aqueous solution of $HAuCl_4$. This was further transferred into 110 °C hot oil bath and stirred for 2 h resulting in a pink-purple solution. To obtain CNF–AuNP precipitate, the solution in oil bath was centrifuged and the resulting pellet was washed multiple times using deionised water.[121]

5.2.2.4 Carbon functionalization. Bacterial CNF also find extensive usage as templates for the synthesis of high stability electrode materials when doped with carbon for various applications including batteries. One such work has been reported by Zhang *et al.*[104] where his group developed carbon nanofiber doped nitrogen electrodes using bacterial CNF/polypyrrole precursor for sodium ion batteries. Briefly, Zhang *et al.* fabricated bacterial CNF precursor by surface assembly of polypyrrole on to the nanofibers *via* polymerisation. The bacterial CNFs were obtained by high speed mechanical homogenisation of bacterial pellicles and deionised water for 20 min. To this suspension, HCl and ferric chloride solution was introduced and sonicated for 15 min. Later, pyrrole solution was added into the mixture at 1–5 °C, mixed well for 15 min and the reaction was allowed to proceed for 5 h. The resulting precipitate was freeze dried under vacuum overnight and finally heated at 700 °C for 3 h in the presence of argon gas.[104] Similarly Chen *et al.*[122] has reported the fabrication of bacterial cellulose derived carbon nanofibers for supercapacitor applications. In his work, bacterial cellulose based carbon nanofibers were doped with heteroatom to enhance the performance of the resulting material. The rectangular pieces of bacterial cellulose pellicles were immersed in aqueous solution of H_3PO_4, $HN_4H_2PO_4$ and H_3BO_3/H_3PO_4 for 10 h at RT under constant stirring followed by liquid nitrogen freezing. This is trailed by freeze drying at −50 °C. The pyrolysis of bacterial cellulose was carried out at 2–520 °C in the presence of N_2 gas for 1 h followed by 5–800 °C for 1 h resulting in the carbon material.[122] Chen *et al.*[123] reported the development of bacterial CNF embedded multi-walled carbon nanotubes (MWCNT) for various medical, electrical and mechanical applications. Initially, bacterial cellulose hydrogels were prepared by

culturing *G. xylinum* BRC5 in HS medium which was later freeze dried and cut in to appropriately sized pieces. This was stored at 25 °C in vacuum oven for 3 h. Later the content was added into 1-allyl-3-methyl-imidazolium chloride solution at 70 °C with constant stirring and introduced with dimethyl sulphoxide solution at RT. To the solution added 0.02% MWCNT under constant stirring for 3 h to obtain a homogenous solution for electrospinning. Finally, CNF with MWCNTs nanocomposite mats were obtained after vacuum drying the electrospun composite membranes washed with ethanol and deionised water.[123]

6. Applications

CNF and its composites from plant and bacterial cellulose has immense applications in various fields like purification of waste water, scaffold substrate for tissue engineering, as drug carrying vehicles for sustained and targeted delivery, cationic and anionic electrodes and as edible coatings for food packaging. A brief account of their application is provided in Table 2. Some of their major applications have been discussed in detail in the following sections.

6.1 Water purification

CNFs of both bacterial and agricultural origin have been used for efficient heavy metal remediation for the conversion of waste water into reusable form. Properties such as high specific surface area, nano size, non-toxicity, hydrophilicity, bio adsorption ability *etc.* makes them suitable candidates to be used as water purifying biomembranes.[124,179,181] Liu *et al.*[124] reported a super-hydrophilic Cu(II) ion adsorbing CNF membrane for the remediation of industrial effluents. In their work, CNF were synthesised *via* TEMPO mediated oxidation of cellulose sludge obtained from agricultural waste and the developed nanofibers were in the size range of 18–40 nm with high specific surface area which was in the range of 134–215 g m^{-2}. They modified the surface of the CNFs using carboxylate groups in the concentration range of 0.6 mmol g^{-1} and 1.5 mmol g^{-1}. They reported that the interaction of surface carboxylate groups with the Cu(II) ions from the effluents leads to the adsorption of Cu ions on to the surface of nanofiber membrane with their gradual and subsequent conversion in to Cu oxide nanoparticles with a size ranging from 200–300 nm. This conversion further enhanced the efficiency and robustness of water purification using the biomembrane. The nanofiber membrane exhibited maximum Cu ion adsorption capacity of 75 mg g^{-1}.[124] Similarly, Meiling *et al.*[125] has reported the fabrication of fluorescent cellulose acetate nanofibers for the detection of Cu^{2+} and Cr^{3+} from polluted water source based on fluorescence. They developed electrospun cellulose acetate nanofibers doped with fluorophore 1,4-dihydroxyanthraquinone (1,4-DHAQ) and demonstrated the changes occurring in the fluorescence intensity of the nanofibers with the adsorption of Cu^{2+} and Cr^{3+}. The fluorescence intensity of the fabricated CNFs were reported to be reducing with increasing adsorption of Cu^{2+} due to the formation of phenolate whereas, increasing concentration of

Cr^{3+} enhanced the fluorescence intensity of the CNF template. The maximum Cu^{2+} adsorption capacity of the nanofiber was reported to be in the range of 2.5×10^{-9} M to 3.75×10^{-8} M.[125]

In another study, Snyder et al.[126] showed the potential of Eucalyptus pulp derived CNF doped with TiO_2, Au and Ag particles in effective removal of organic compound, methylene blue from contaminated water source. They demonstrated that the photocatalytic activity of TiO_2–CNF enhanced with the surface functionalization of Au and Ag particles in the presence of simulated sunlight.[117] Cellulose acetate nanofibers modified using oxolane-2,5-dione has also been reported as excellent adsorbents of heavy metal ions such as lead and cadmium from contaminated water sources by Stephen et al.[127] The developed CNF composite exhibited high reusable capacity without compromising its adsorption efficiency.[127] CNF based aerogels has also been successfully employed in decontamination of water. Xu He et al.[128] demonstrated the efficiency of chemically crosslinked, highly porous CNF aerogels functionalised with ammonium in remediating Cr(IV) ions from waste water. They reported that 99% of Cr(IV) ions from 1 L of contaminated water was successfully remediated using 1 g of CNF aerogel.[128] In another study, Sehaqui et al.[92] used waste pulp derived CNF for efficient removal of PO_4^{3-}, SO_4^{2-}, F^- and NO^{3-} from polluted water source. They demonstrated the transformation of CNFs to cationic CNFs by the surface functionalization of quaternary ammonium. The developed CNF paper showed high potential towards nitrate removal (380 mg m^{-2}) when compared to other anions present in the contaminated water.[92] Similar to the plant derived and modified CNF, bacterial originated CNFs also acted as potential bioadsorbant agents in water purification. Bhavna, Mohite and Patil[95] reported that cellulose obtained from Gluconacetobacter hansenii successfully removed heavy metal ions such as Pb^{2+}, Cd^{2+} and Ni^{2+}. They also showed the potential of bacterial CNF in removal of azo dyes resulting in clean environment.[95] Similarly, Wang et al.[105] reported the fabrication of Cu^{2+} and Pb^{2+} adsorbing biomembranes made of bacterial CNFs coated with polyethylenimine using flush coating technique for purification of waste water. The fabricated membrane exhibited superior adsorption and desorption capacity after acid treatment using ethylene diamine tetra acetic acid and better recycling ability. A unique feature of this biomembrane is reported to be their ability to adsorb Cu^{2+} ions, their subsequent reduction into copper nanoparticles which further catalyses the reduction of methylene blue dye from aqueous solutions.[110] CNFs were also used to separate oil and organic contaminants from water. One such study was reported by Huazheng Sai et al.[129] where CNFs derived from bacteria was used as oil adsorbents from water sources. They used a method in which the nanofiber was surface modified using trimethylchlorosilane to obtain CNFs with hydrophobicity. The resulting nanofibers exhibited well defined porous structure with high surface area and oil absorption ability of 185 g g^{-1}.[129]

6.2   Tissue engineering and drug delivery

CNFs also find extensive applications in the area of bone, cartilage and skin tissue engineering due to its large surface area, tuneable pore size, non-toxicity, biocompatibility, biodegradability and superior mechanical strength. In addition, their ability to deliver the bioactive compounds in sustained manner along with tuneable surface properties spikes their demand as effective drug carriers in targeted drug delivery. In a study carried out by Nasri-Nasrabadi et al.[130] starch/cellulose composite nanofibers were used to fabricate porous scaffolds for cartilage tissue engineering. In this method, NaCl was used as the porogen to develop suitable pore size in the nanofibers which impart enhanced surface properties to the scaffold. The incorporation of higher content of CNF was reported to enhance the tensile strength, Young's modulus and hydrophilicity of the resulting scaffold. They also reported slow degradation of the fabricated scaffold in vitro (>20 weeks) along with enhanced rabbit chondrocyte attachment and proliferation.[130] Likewise, CNFs derived from bacterial sources are also used to develop scaffolds as substrates suitable for cell attachment and proliferation. One such bacterial CNF based porous scaffold was developed by Zhang et al.[131] They reported A. xylinum X-2 extracted CNF surface immobilised with lecithin which was further crosslinked with proanthocyanidin. The developed scaffold exhibited superior mechanical strength, hydrophilicity, cytocompatibility, porosity and thermal stability.[131] Modified cellulose such as cellulose acetate, hydroxypropyl cellulose and hydroxymethyl cellulose also exhibited remarkable potential to be used in tissue engineering. Chahal et al.[132] fabricated modified cellulose–PVA blended electrospun nanofiber based scaffold for bone tissue engineering. High mechanical properties, stability, bead free morphology of the nanofiber, biocompatibility and biodegradability are some of the highlighting features of this scaffold.[132] Qiao et al.[133] developed highly porous composite hydrogels made of bacterial CNFs and PVA and studied their mechanical properties to understand their suitability for various biomedical applications. Their high crystallinity, superior tensile strength and Young's modulus makes them highly recommended substrate for wound dressing and for the synthesis of artificial blood vessels and cartilage.[133]

Modified CNFs also acted as suitable drug carriers. Hydrophobic drugs such as sulfisoxazole encapsulated hydroxy propyl-β-cyclodextrin incorporated hydroxy propyl CNFs were fabricated to develop sustainable drug delivery system.[134] The release of sulfisoxazole from the developed nanofiber material was prolonged up to 720 min. High surface area, enhanced solubility of the drug, controlled drug delivery and biocompatibility makes them a potential choice as wound dressing material. Similarly, Opanasopit et al.[135] reported the fabrication of transdermal patches containing cellulose acetate and PVA nanofibers incorporated with capsicum extract. The developed material showed high skin permeation while tested on shed snake skin as the model and superior cytocompatibility while studied on normal human foreskin fibroblast cells (NHF).[135]

6.3   Food packaging

CNFs are used as edible coatings to preserve the shelf life of various fruits and vegetables. They carry out this function by reducing the microbial growth on the fruits and vegetables,

oxidation and by reducing the transfer of moisture thereby maintaining the quality of the products.[136] Andrade et al.[136] reported a study on the viscosity and spreading dynamics of the edible coating formulation developed using CNFs along with glycerol and gelatin on the surface of banana and eggplant epicarps. They reported the development of drop method using a high speed syringe pump to produce the coating on the surface of the fruits. The CNF used for this application was obtained from *Gluconacetobacter swingsii* and pineapple peel juice. They prepared the formulation by initially hydrating gelatin in 100 mL of distilled water at RT for 30 min in the concentration range of 0.6, 1.3 and 2 g. After complete dissolution of gelatin, CNF and glycerol was added in the concentration range of 1 g, 5 g and 10 g, 20 g per 100 g of gelatin respectively. Finally, the mix was sonicated for 30 min. SitiHajar Othman,[137] Vijayendra and Shamala,[138] Tang et al.[139] and Babak Ghanbarzadeh et al.[140] has reviewed in detail the role of CNF along with other biopolymers as nanofillers and reinforcing agent to enhance the mechanical properties, duration and stability of the resulting coating material. Panaitescu et al.[141] also reported the use of CNF based materials for food packaging. In his work, CNF obtained from microcrystalline cellulose was used as reinforcing agent in starch/PVA composite films, which was further tested for its efficacy as food packaging films. It was reported that addition of CNF enhanced mechanical strength and stability of the composite polymer films.[141]

CNF are also used as reinforcing agents in food packaging material. The applications of food packaging materials available commercially are limited due to its poor mechanical properties. Incorporation of CNF enhances the efficiency of the fabricated edible materials for packing the foods. The potential of CNFs as reinforcing agents in developing edible coatings for fruits and vegetables are reported by Azeredo et al.[142] They tried to develop edible films of mango puree incorporated with CNFs. The presence of cellulose fibers are reported to enhance the mechanical properties such as tensile strength and elastic modulus of the fabricated film. The cellulose fibers also reduced the permeability of water vapour thereby ensuring high stability of the composite edible film produced.[142]

6.4 Miscellaneous applications

CNFs find extensive application as energy storage devices. Yang et al.[143] reported the fabrication of super capacitor electrodes using CNF and multi walled carbon nanotubes (MWCNT) surface coated using polyaniline. The resulting porous aerogel capacitor electrode exhibit minimum charge transfer resistance, super specific capacitance, flexibility and are cost effective.[143] Similarly, Wang and Li[103] developed foldable hybrid supercapacitor electrode using CNF coated with MWCNTs and TiO_2 nanotubes. The resulting electrode exhibited high mechanical strength, flexibility, enhanced discharge capacity and superior cyclability.[103] Bacterial CNF based electrode has been fabricated for sodium ion batteries by Zhang et al.[104] Superior reversible specific capacity, high stability and excellent electrochemical activities are outstanding features of this electrode.[104] CNFs are also used as agents to identify adulterants in

cosmetics and pharmaceutical products. Tidjarat et al.[144] reported the use of cellulose acetate nanofibers as screening agents to identify the adulterations in retinoic acid and hydroquinone present in the cosmetic products using chromatographic technique.[144] A similar study using cellulose acetate was also reported by Rojanarata et al.[145] in identifying the adulterations in steroid present in various nutraceuticals and pharmaceuticals.[145]

7. Conclusions

This review presents various techniques employed for the extraction of CNF from plants and bacterial sources. The study suggests that the extraction of homogenous CNFs from different plant sources such as rice, wheat straw, wood, banana, cotton, sea grasses *etc.* are feasible but accompanied with certain drawbacks. It is critical to choose an apt treatment method for their isolation. It is important to employ an eco-friendly extraction technique that results in high mechanical strength and integrity of the resulting CNF along with purity, biocompatibility, crystallinity, wettability and surface tuneable structure. The properties of CNF are very unique and specific to the extraction method employed thereby making them complementary to each other. In case of extracting CNF from plant sources, a combination of mechanical, physical, chemical and enzymatic treatment results in better yield than adopting a single treatment option. The use of mechanical disintegration techniques such as homogeniser and microfluidizer are limited due to disadvantages like fiber clogging and requirement of high energy source. Thus for the high scale production of CNF, oxidation of fibers prior to their exposure to mechanical treatment, results in easier defibrillation. The isolation of CNF from bacterial sources is comparatively easier and results in high purity CNF. However, the duration required for the production of complete CNF limits their benefits in the industrial scale. Thus there is a need to develop a technique that provides high yield of pure CNF within short span of time. In addition to this, the properties of CNF specific for particular applications can be customised by blending them with various hydrophilic and hydrophobic polymers. Surface doping of CNF with metals, non-metals and numerous nanoparticles are also addressed in this review. To recapitulate, this review converges the different extraction strategies employed for the isolation of CNF from various plant and bacterial species.

Conflicts of interest

There are no conflicts to declare.

Abbreviations

CNF	Cellulose nanofiber
CNC	Cellulose nanocrystal
hDFC	Human dental follicle cells
RhB	Rhodamine B
TFNC	Thin film nanofibrous composite

DTPA	Diethylene triamine penta acetic acid
TEMPO	2,2,6,6-Tetramethylpiperidin-1-yl)oxy radical oxidation
NaOH	Sodium hydroxide
KOH	Potassium hydroxide
FDP	Flat panel display
DHAQ	Dihydroxy anthraquinone
HAp	Hydroxyapatite
AOBC	Amidoxime surface functionalised CNF
MWCNT	Multi wall carbon nanotube

Acknowledgements

References

1 D. Klemm, B. Heublein, H. P. Fink and A. Bohn, *Angew. Chem., Int. Ed.*, 2005, **44**, 3358–3359.

2 H. A. Khalil, A. H. Bhat and A. I. Yusra, *Carbohydr. Polym.*, 2012, **87**, 963–979.

3 I. N. Siró and D. Plackett, *Cellulose*, 2010, **17**, 459–494.

4 J. Bras, D. Viet, C. Bruzzese and A. Dufresne, *Carbohydr. Polym.*, 2011, **84**, 211–215.

5 R. J. Moon, A. Martini, J. Nairn, J. Simonsen and J. Youngblood, *Chem. Soc. Rev.*, 2011, **40**, 3941–3994.

6 E. M. Teixeira, A. C. Corrêa, A. Manzoli, A. F. L. Leite, C. R. Oliveira and L. H. Mattoso, *Cellulose*, 2010, **17**, 595–606.

7 D. N. Saheb and J. P. Jog, *Adv. Polym. Technol.*, 1999, **18**, 351–363.

8 S. T. Georgopoulos, P. A. Tarantili, E. Avgerinos, A. G. Andreopoulos and E. G. Koukios, *Polym. Degrad. Stab.*, 2005, **90**, 303–312.

9 T. Subbiah, G. S. Bhat, R. W. Tock, S. Parameswaran and S. S. Ramkumar, *J. Appl. Polym. Sci.*, 2005, **96**, 557–569.

10 L. Yang, Z. Hong-Yu, Q. Yang and L. Da-nian, *J. Appl. Polym. Sci.*, 2012, **126**, 244–250.

11 A. Sato, D. Kabusaki, H. Okumura, T. Nakatani, F. Nakatsubo and H. Yano, *Composites, Part A*, 2016, **83**, 72–79.

12 L. Heath and W. Thielemans, *Green Chem.*, 2010, **12**, 1448–1453.

13 S. Beck-Candanedo, M. Roman and D. G. Gray, *Biomacromolecules*, 2005, **6**, 1048–1054.

14 M. Samyn, P. Laborie, A. P. Mathew, A. Airoudj, H. Haidara and V. Roucoules, *Langmuir*, 2012, **28**, 1427–1438.

15 M. Pääkkö, M. Ankerfors, H. Kosonen, A. Nykänen, S. Ahola, M. Österberg, J. Ruokolainen, J. Laine, P. T. Larsson, O. Ikkala and T. Lindström, *Biomacromolecules*, 2007, **8**, 1934–1941.

16 P. Podsiadlo, S. Y. Choi, B. Shim, J. Lee, M. Cuddihy and N. A. Kotov, *Biomacromolecules*, 2005, **6**, 2914–2918.

17 B. Soni, E. B. Hassan and B. Mahmoud, *Carbohydr. Polym.*, 2015, **134**, 581–589.

18 G. F. Torres, S. Commeaux and O. P. Troncoso, *J. Funct. Biomater.*, 2012, **3**(4), 864–878.

19 M. Jonoobi, *et al.*, *Compos. Sci. Technol.*, 2010, **70**(12), 1742–1747.

20 A. N. Frone, *et al.*, *J. Mater. Sci.*, 2016, **51**(21), 9771–9791.

21 N. Najib and C. Christodoulatos, *Proceedings of the 13th International Conference of Environmental Science and Technology*, Athens, Greece, 2013.

22 M. Labet and W. Thielemans, *Cellulose*, 2011, **18**(3), 607–617.

23 Y. Luo, J. Zhang, X. Li, C. Liao and X. Li, *J. Nanomater.*, 2014, **2014**, 11.

24 J. I. Morán, V. A. Alvarez, V. P. Cyras and A. Vázquez, *Cellulose*, 2008, **15**, 149–159.

25 W. Chen, Q. Li, J. Cao, Y. Liu, J. Li, J. Zhang, S. Luo and H. Yu, *Carbohydr. Polym.*, 2015, **117**, 950–956.

26 A. N. Nakagaito and H. Yano, *Appl. Phys. A: Mater. Sci. Process.*, 2004, **78**, 547–552.

27 I. M. Saito, C. Cabello and R. S. Fukushima, *Revista Raízes e Amidos Tropicais*, 2006, **2**, 1–11.

28 K. Ohkawa, *Molecules*, 2015, **20**, 9139–9154.

29 R. Dashtbani and E. Afra, *Int. J. Nano Dimens.*, 2015, **6**, 1.

30 J. S. Eichhorn, *et al.*, *J. Mater. Sci.*, 2010, **45**(1), 1.

31 C. C. Jose, L. Mathew and S. Thomas, *Rev. Adv. Mater. Sci.*, 2014, **37**, 1.

32 W. Haoran, *et al.*, *Environ. Sci.: Nano*, 2014, **1**(4), 302–316.

33 O. Nechyporchuk, M. N. Belgacem and J. Bras, *Ind. Crops Prod.*, 2016, **93**, 2–25.

34 M. Albert, *J. Appl. Polym. Sci.*, 2011, **119**(4), 2449–2460.

35 J. Olivier, *et al.*, *Nature*, 2000, **408**, 157.

36 N. Shahid, D. R. Hahn and G. Schuurman, *Nature*, 2000, **403**, 762.

37 N. W. Scott, *Ambio*, 1990, 101.

38 E. Nicolai and R. D. Preston, *Proc. R. Soc. London, Ser. B*, 1952, **140**(899), 244–274.

39 M. Pääkkö, J. Vapaavuori, R. Silvennoinen, H. Kosonen, M. Ankerfors, T. Lindström, L. A. Berglund and O. Ikkala, *Soft Matter*, 2008, **4**(12), 2492–2499.

40 H. Noriko, T. Kondo and M. Ishihara, *Carbohydr. Polym.*, 2005, **61**(2), 191–197.

41 R. Ek, C. Gustafsson, A. Nutt, T. Iversen and C. Nyström, *J. Mol. Recognit.*, 1998, **11**(1–6), 263–265.

42 A. Mihranyan, A. P. Llagostera, R. Karmhag, M. Strømme and R. Ek, *Int. J. Pharm.*, 2004, **269**(2), 433–442.

43 M. Johnson, S. Shivkumar and L. Berlowitz-Tarrant, *Mater. Sci. Eng., B*, 1996, **38**(1), 103–108.

44 E. A. Bayer, H. Chanzy, R. Lamed and Y. Shoham, *Curr. Opin. Struct. Biol.*, 1998, **8**(5), 548–557.

45 C. O'sullivan Antoinette, *Cellulose*, 1997, **4**(3), 173–207.

46 T. Ioannes, *J. Phycol.*, 1999, **35**(4), 635–655.

47 N. Hayashi, J. Sugiyama, T. Okano and M. Ishihara, *Carbohydr. Res.*, 1997, **305**(1), 109–116.

48 M. Albert, K. Edsman and M. Strømme, *Food Hydrocolloids*, 2007, **21**(2), 267–272.

49 G. Nyström, A. Razaq, M. Strømme, L. Nyholm and A. Mihranyan, *Nano Lett.*, 2009, **9**(10), 3635–3639.

50 M. M. Andrade-Mahecha, F. M. Pelissari, D. R. Tapia-Blácido and F. C. Menegalli, *Carbohydr. Polym.*, 2015, **123**, 406–415.

51 I. K. Neelamana, S. Thomas and J. Parameswaranpillai, *J. Appl. Polym. Sci.*, 2013, **130**, 1239–1246.

52 E. M. Teixeira, A. C. Corrêa, A. Manzoli, A. F. L. Leite, C. R. Oliveira and L. H. Mattoso, *Cellulose*, 2010, **17**, 595–606.

53 F. Bettaieb, O. Nechyporchuk, R. Khiari, M. F. Mhenni, A. Dufresne and M. N. Belgacem, *Carbohydr. Polym.*, 2015, **134**, 664–672.

54 X. He, Q. Xiao, C. Lu, Y. Wang, X. Zhang, J. Zhao, W. Zhang, X. Zhang and Y. Deng, *Biomacromolecules*, 2014, **15**, 618–627.

55 X. Li, D. Yuan, J. Zhang, Z. Lin and X. Zhang, *PLoS One*, 2013, **8**, e54444.

56 Q. Liu, J. Li, Y. Zhao, Y. Zhou and C. Li, *Mater. Lett.*, 2015, **138**, 89–91.

57 P. Preethi and G. B. Murthy, *Agrotechnology*, 2013, **8**(S11), 2.

58 S. F. Souza, A. L. Leao, J. H. Cai, C. Wu, M. Sain and B. M. Cherian, *Mol. Cryst. Liq. Cryst.*, 2010, **522**, 42–342.

59 S. Hooshmand, Y. Aitomäki, N. Norberg, A. P. Mathew and K. Oksman, *ACS Appl. Mater. Interfaces*, 2015, **7**, 13022–13028.

60 P. Khawas and S. C. Deka, *Carbohydr. Polym.*, 2015, 1–26.

61 W. Chen, H. Yu, Y. Liu, P. Chen, M. Zhang and Y. Hai, *Carbohydr. Polym.*, 2011, **83**, 1804–1811.

62 P. C. Tischer, M. R. Sierakowski, H. Westfahl Jr and C. A. Tischer, *Biomacromolecules*, 2010, **11**, 1217–1224.

63 F. M. Pelissari, M. M. Andrade-Mahecha, P. J. Sobral and F. C. Menegalli, *Starch/Staerke*, 2012, **64**, 382–391.

64 H. Tibolla, F. M. Pelissari and F. C. Menegalli, *LWT–Food Sci. Technol.*, 2014, **59**, 1311–1318.

65 H. Sehaqui, S. Morimune, T. Nishino and L. A. Berglund, *Biomacromolecules*, 2012, **13**, 3661–3667.

66 T. Saito, R. Kuramae, J. Wohlert, L. A. Berglund and A. Isogai, *Biomacromolecules*, 2013, **14**, 248–253.

67 M. Bergenstråhle, L. A. Berglund and K. Mazeau, *J. Phys. Chem. B*, 2007, **111**, 9138–9145.

68 N. Butchosa and Q. Zhou, *Cellulose*, 2014, **21**, 4349–4358.

69 A. Kentaro, S. Iwamoto and H. Yano, *Biomacromolecules*, 2007, **8**(10), 3276–3278.

70 W. Stelte and A. R. Sanadi, *Ind. Eng. Chem. Res.*, 2009, **48**, 11211–11219.

71 K. Abe, S. Iwamoto and H. Yano, *Biomacromolecules*, 2007, **8**, 3276–3278.

72 H. Ma, B. Zhou, H. S. Li, Y. Q. Li and S. Y. Ou, *Carbohydr. Polym.*, 2011, **84**, 383–389.

73 H. Yano, S. Sasaki, M. Shams and K. Abe, *Adv. Opt. Mater.*, 2014, **2**, 231–234.

74 C. Terenzi, K. Prakobna, L. A. Berglund and L. Furó, *Biomacromolecules*, 2015, **16**, 1506–1515.

75 S. Galland, Y. Leterrier, T. Nardi, C. J. Plummer, J. A. Månson and L. A. Berglund, *J. Appl. Polym. Sci.*, 2014, **131**, 16.

76 G. Gellerstedt and I. Pettersson, *J. Wood Chem. Technol.*, 1982, **2**, 231–250.

77 X. Z. Zhang, R. C. Francis, D. B. Dutton and R. T. Hill, *J. Wood Chem. Technol.*, 1998, **18**, 253–266.

78 A. S. Jaaskelainen, T. Tapanila and K. Poppius-Levlin, *J. Wood Chem. Technol.*, 2000, **20**, 43–59.

79 J. Zhao, W. Zhang, W. Zhang, X. Zhang, C. Lu and Y. Deng, *Carbohydr. Polym.*, 2013, **97**, 695–702.

80 F. Ansari, M. Skrifvars and L. Berglund, *Compos. Sci. Technol.*, 2015, **117**, 298–306.

81 B. Atanu, B. C. Saha, J. W. Lawton, R. L. Shogren and J. L. Willett, *Carbohydr. Polym.*, 2006, **64**, 134–137.

82 J. O. Wiiiam, S. Justin, H. I. Syed, M. G. Gregory, E. G. Mara and F. R. Jean, *J. Polym. Environ.*, 2005, **13**, 301–306.

83 Z. Xu, Q. H. Wang, Z. Jiang, X. X. Yang and Y. Z. Ji, *Biomass Bioenergy*, 2007, **31**, 162–167.

84 J. Zhao, X. Z. Li and Y. B. Qu, *Bioresour. Technol.*, 2006, **97**, 1470–1476.

85 C. Cara, E. Ruiz, M. Ballesteros, P. Manzanares, M. J. Negro and E. Castro, *Fuel*, 2008, **87**, 692–700.

86 A. C. Leung, E. Lam, J. Chong, S. Hrapovic and J. H. Luong, *J. Nanopart. Res.*, 2013, **15**, 1–24.

87 E. Kalita, B. K. Nath, P. Deb, F. Agan, R. Md. Islam and K. Saikia, *Carbohydr. Polym.*, 2015, **122**, 308–313.

88 B. Nasri-Nasrabadi, M. Mehrasa, M. Rafienia, S. Bonakdar, T. Behzad and S. Gavanji, *Carbohydr. Polym.*, 2014, **108**, 232–238.

89 A. Alemdar and M. Sain, *Compos. Sci. Technol.*, 2008, **68**, 557–565.

90 S. Elanthikkal, U. Gopalakrishnapanicker, S. Varghese and J. T. Guthrie, *Carbohydr. Polym.*, 2010, **80**, 852–859.

91 H. B. Bettaieb, M. Esguerra, D. Delbro, B. Risberg and P. Gatenholm, *J. Tissue Eng. Regener. Med.*, 2008, **2**, 320–330.

92 H. Sehaqui, A. Mautner, U. P. Larraya, N. Pfenninger, P. Tingaut and T. Zimmermann, *Carbohydr. Polym.*, 2016, **136**, 334–340.

93 D. Theng, G. Arbat, M. Delgado-Aguilar, F. Vilaseca, B. Ngo and P. Mutjé, *Ind. Crops Prod.*, 2015, **76**, 166–173.

94 K. Y. Lee, G. Buldum, A. Mantalaris and A. Bismarck, *Macromol. Biosci.*, 2014, **14**, 10–32.

95 B. V. Mohite and S. V. Patil, *J. Biomater. Sci., Polym. Ed.*, 2014, **25**, 2053–2065.

96 J. Fang, S. Kawano, K. Tajima and T. Kondo, *Biomacromolecules*, 2015, **16**, 3154–3160.

97 S. Gonçalves, J. Padrão, I. P. Rodrigues, J. P. Silva, V. Sencadas, S. Lanceros-Mendez, H. Girão, F. Dourado and L. R. Rodrigues, *Biomacromolecules*, 2015, **16**, 1341–1351.

98 F. Quero, M. Nogi, K. Y. Lee, G. V. Poel, A. Bismarck, A. Mantalaris, H. Yano and S. J. Eichhorn, *ACS Appl. Mater. Interfaces*, 2010, **3**, 490–499.

99 R. T. Olsson, R. Kraemer, A. Lopez-Rubio, S. Torres-Giner, M. J. Ocio and J. M. Lagarón, *Macromolecules*, 2010, **43**, 4201–4209.

100 K. Y. Lee, J. J. Blaker and A. Bismarck, *Compos. Sci. Technol.*, 2009, **69**, 2724–2733.

101 H. Bäckdahl, M. Esguerra, D. Delbro, B. Risberg and P. Gatenholm, *J. Tissue Eng. Regener. Med.*, 2008, **2**, 320–330.

102 C. Yang and D. Li, *Mater. Lett.*, 2015, **155**, 78–81.

103 F. Wang and D. Li, *Mater. Lett.*, 2015, **158**, 119–122.

104 Z. Zhang, J. Zhang, X. Zhao and F. Yang, *Carbon*, 2005, **95**, 552–559.

105 J. Wang, X. Lu, P. F. Ng, K. I. Lee, B. Fei, J. H. Xin and W. Jian-yong, *J. Colloid Interface Sci.*, 2015, **440**, 32–38.

106 J. Fu, D. Li, G. Li, F. Huang and Q. Wei, *J. Electroanal. Chem.*, 2015, **738**, 92–99.

107 Y. Jiazhi, L. Xiaoli, H. Liyonga and S. Dongping, *Chin. J. Chem. Eng.*, 2013, **21**, 1419–1424.

108 M. Gopiraman, H. Bang, G. Yuan, C. Yin, S. Kyung-Hun, J. S. Lee, I. M. Chung, R. Karvembu and I. S. Kim, *Carbohydr. Polym.*, 2015, **132**, 554–564.

109 L. Tian, S. Zi-qiang, W. Jian-quan and G. Mu-jia, *Chem. Eng. J.*, 2015, **260**, 818–825.

110 J. Wang, X. Lu, P. F. Ng, K. I. Lee, B. Fei, J. H. Xin and W. Jian-yong, *J. Colloid Interface Sci.*, 2015, **440**, 32–38.

111 H. Wang, E. Zhu, J. Yang, P. Zhou, D. Sun and W. Tang, *J. Phys. Chem. C*, 2012, **116**, 13013–13019.

112 W. Hu, S. Chen, Z. Yang, L. Liu and H. Wang, *J. Phys. Chem. B*, 2011, **115**, 8453–8457.

113 Z. Lin, Z. Guan and Z. Huang, *Ind. Eng. Chem. Res.*, 2013, **52**, 2869–2874.

114 R. Hobzova, M. Duskova-Smrckova, J. Michalek, E. Karpushkin and P. Gatenholm, *Polym. Int.*, 2012, **61**, 1193–1201.

115 M. Nogi and H. Yano, *Adv. Mater.*, 2008, **20**, 1849–1852.

116 F. Kramer, D. Klemm, D. Schumann, N. Hebler, F. Wesarg, W. Fried and D. Stadermann, *Macromol. Symp.*, 2006, **244**, 136–148.

117 L. E. Millon and W. K. Wan, *J. Biomed. Mater. Res., Part B*, 2006, **79**, 245–253.

118 E. E. Brown and M. P. Laborie, *Biomacromolecules*, 2007, **8**, 3074–3081.

119 P. Zhou, H. Wang, J. Yang, J. Tang, D. Sun and W. Tang, *Ind. Eng. Chem. Res.*, 2012, **51**, 5743–5748.

120 S. Ifuku, M. Tsuji, M. Morimoto, H. Saimoto and H. Yano, *Biomacromolecules*, 2009, **10**, 2714–2717.

121 M. Chen, H. Kang, Y. Gong, J. Guo, H. Zhang and R. Liu, *ACS Appl. Mater. Interfaces*, 2015, **7**, 21717–21726.

122 L. F. Chen, Z. H. Huang, H. W. Liang, H. L. Gao and S. H. Yu, *Adv. Funct. Mater.*, 2014, **24**, 5104–5111.

123 P. Chen, Y. S. Yun, H. Bak, S. Y. Cho and H. J. Jin, *Mol. Cryst. Liq. Cryst.*, 2010, **519**, 69–78.

124 P. Liu, K. Oksman and A. P. Mathew, *J. Colloid Interface Sci.*, 2016, **464**, 175–182.

125 W. Meiling, G. Meng, Q. Huang and Y. Qian, *Environ. Sci. Technol.*, 2012, **46**, 367–373.

126 M. Snyder, Z. Bo, R. Moon, R. Jean-Christophe and L. Stanciu, *J. Colloid Interface Sci.*, 2013, **399**, 92–98.

127 M. Stephen, N. Catherine, M. Brenda, K. Andrew, P. Leslie and G. Corrine, *J. Hazard. Mater.*, 2011, **192**, 922–927.

128 X. He, L. Cheng, Y. Wang, J. Zhao, W. Zhang and C. Lu, *Carbohydr. Polym.*, 2014, **111**, 683–687.

129 H. Sai, L. Xing, J. Xiang, L. Cui, J. Jiao, C. Zhao, Z. Li and F. Li, *J. Mater. Chem. A*, 2013, **1**, 7963–7970.

130 B. Nasri-Nasrabadi, M. Mehrasa, M. Rafienia, S. Bonakdar, T. Behzad and S. Gavanji, *Carbohydr. Polym.*, 2014, **108**, 232–238.

131 J. Zhang, P. Chang, C. Zhang, G. Xiong, H. Luo, Y. Zhu, K. Ren, F. Yao and Y. Wan, *React. Funct. Polym.*, 2015, **91–92**, 100–107.

132 S. Chahal, F. S. Hussain and M. B. Yusoff, *Procedia Eng.*, 2013, **53**, 683–688.

133 K. Qiao, Y. Zheng, S. Guo, J. Tan, X. Chen, J. Li, D. Xu and J. Wang, *Compos. Sci. Technol.*, 2015, **118**, 47–54.

134 Z. Aytac, H. S. Sen, E. Durgun and T. Uyar, *Colloids Surf., B*, 2015, **128**, 331–338.

135 P. Opanasopit, W. Sila-on, T. Rojanarata and T. Ngawhirunpat, *Pharm. Dev. Technol.*, 2013, **18**, 1140–1147.

136 R. Andrade, O. Skurtys and F. Osorio, *LWT–Food Sci. Technol.*, 2015, **61**, 422–429.

137 S. H. Othman, *Agric. Agric. Sci. Procedia*, 2014, **2**, 296–303.

138 S. V. Vijayendra and T. R. Shamala, *Crit. Rev. Biotechnol.*, 2014, 1–20.

139 X. Z. Tang, P. Kumar, S. Alavi and K. P. Sandeep, *Crit. Rev. Food Sci. Nutr.*, 2012, **52**, 426–442.

140 B. Ghanbarzadeh, S. A. Oleyaei and H. Almasi, *Crit. Rev. Food Sci. Nutr.*, 2015, **55**, 1699–1723.

141 D. M. Panaitescu, A. N. Frone, M. Ghiurea and L. Chiulan, *Ind. Crops Prod.*, 2015, **70**, 170–177.

142 H. M. De Azeredo, *Food Res. Int.*, 2009, **42**, 1240–1253.

143 C. Yang, C. Chen, Y. Pan, S. Li, F. Wang, J. Li, N. Li, X. Li, Y. Zhang and D. Li, *Electrochim. Acta*, 2015, **182**, 264–271.

144 S. Tidjarat, W. Winotapun, P. Opanasopit, T. Ngawhirunpat and T. Rojanarata, *J. Chromatogr. A*, 2014, **1367**, 141–147.

145 T. Rojanarata, S. Plianwong, K. Su-uta, P. Opanasopit and T. Ngawhirunpat, *Talanta*, 2013, **115**, 208–213.

146 F. Jiang and H. You-Lo, *Carbohydr. Polym.*, 2015, **122**, 60–68.

147 H. Yousefi, T. Nishino, M. Faezipour, G. Ebrahimi and A. Shakeri, *Biomacromolecules*, 2011, **12**, 4080–4085.

148 E. Tamahkar, T. Kutsal and A. Denizli, *Process Biochem.*, 2015, **50**(12), 2289–2297.

149 S. Bottan, F. Robotti, P. Jayathissa, A. Hegglin, N. Bahamonde, J. A. Heredia-Guerrero, I. S. Bayer, A. Scarpellini, H. Merker, N. Lindenblatt, D. Poulikakos and A. Ferrari, *ACS Nano*, 2015, **9**, 206–219.

150 L. Liu, D. He, W. Guang-Sheng and Y. Shu-Hong, *Langmuir*, 2011, **27**, 7199–7206.

151 M. Schreiber, S. Vivekanandhan, A. K. Mohanty and M. Misra, *ACS Sustainable Chem. Eng.*, 2015, **3**, 33–41.

152 T. Stylianopoulos, M. Kokonou, S. Michael, A. Tryfonos, C. Rebholz, A. D. Odysseos and C. Doumanidis, *J. Biomed. Mater. Res., Part B*, 2012, **100**, 2222–2230.

153 A. K. Bledzki, A. A. Mamun, M. Lucka-Gabor and V. S. Gutowski, *eXPRESS Polym. Lett.*, 2008, **2**, 413–422.

154 G. L. Louis and B. A. K. Andrews, *Text. Res. J.*, 1987, **57**, 339–345.

155 A. Thygesen, J. Oddershede, H. Lilholt, A. B. Thompsen and K. Stahl, *Cellulose*, 2005, **12**, 563–576.

156 J. D. Megiatto Jr, Fibras de Sisal: Estudo de propriedades e modificações químicas visando a aplicação em compósitos

de matriz fenólica, Ph.D. thesis, Federal University of São Carlos, Brazil, 2006.

157 K. Devarayan and K. Byoung-Suhk, *Sens. Actuators, B*, 2015, **209**, 281–286.

158 J. Matulevicius, L. Kliucininkas, T. Prasauskas, D. Buivydiene and D. Martuzevicius, *J. Aerosol Sci.*, 2016, **92**, 27–37.

159 C. Zhijiang, X. Yi, Y. Haizheng, J. Jia and Y. Liu, *Mater. Sci. Eng., C*, 2016, **58**, 757–767.

160 F. A. Aouada, M. R. De Moura, W. J. Orts and L. H. C. Mattoso, *J. Agric. Food Chem.*, 2011, **59**, 9433–9442.

161 S. M. Musyoka, J. C. Ngila and B. B. Mamba, *Phys. Chem. Earth*, 2013, **66**, 45–50.

162 V. Kuzmenko, O. Naboka, H. Staaf, M. Haque, G. Goransson, P. Lundgren, P. Gatenholm and P. Enoksson, *Mater. Chem. Phys.*, 2015, **160**, 59–65.

163 M. Wang, G. Meng, Q. Huang and Y. Qian, *Environ. Sci. Technol.*, 2012, **46**, 367–373.

164 F. Ahmed, S. Saleemi, Z. Khatri, M. I. Abro and K. Ick-Soo, *Carbohydr. Polym.*, 2015, **115**, 388–393.

165 P. P. S. Lacerda, A. M. M. V. Barros-Timmons, C. S. R. Freire, A. J. D. Silvestre and C. P. Neto, *Biomacromolecules*, 2013, **14**, 2063–2073.

166 W. D. Ding, R. K. M. Chu, L. H. Mark, C. B. Park and M. Sain, *Eur. Polym. J.*, 2015, **71**, 231–247.

167 K. Benhamou, H. Kaddami, A. Magnin, A. Dufresne and A. Ahmad, *Carbohydr. Polym.*, 2015, **122**, 202–211.

168 K. Prakobna, C. Terenzi, Q. Zhou, I. Furó and L. A. Berglund, *Carbohydr. Polym.*, 2015, **125**, 92–102.

169 M. Babaee, M. Jonoobi, Y. Hamzeh and A. Ashori, *Carbohydr. Polym.*, 2015, **132**, 1–8.

170 A. Lavoratti, L. C. Scienza and A. J. Zattera, *Carbohydr. Polym.*, 2016, **136**, 955–963.

171 K. Prakobna, S. Galland and L. A. Berglund, *Biomacromolecules*, 2015, **16**, 904–912.

172 E. Tamahkar, T. Kutsal and A. Denizli, *Process Biochem.*, 2015, **50**(12), 2289–2297.

173 N. Volk, R. He and R. Magniez, *Eur. Polym. J.*, 2015, **72**, 270–281.

174 A. N. Frone, C. A. Nicolae, R. A. Gabor and D. Panaitescu, *Polym. Degrad. Stab.*, 2015, **121**, 385–397.

175 A. Abdulkhani, J. Hosseinzadeh, A. Ashori, S. Dadashi and Z. Takzare, *Polym. Test.*, 2014, **35**, 73–79.

176 H. Almasi, B. Ghanbarzadeh, J. Dehghannya, A. A. Entezami and A. K. Asl, *Food Packaging and Shelf Life*, 2015, **5**, 21–31.

177 C. Zhou, R. Chu, R. Wu and Q. Wu, *Biomacromolecules*, 2011, **12**, 2617–2625.

178 N. Chitpong and M. S. Husson, *J. Membr. Sci.*, 2017, **523**, 418–429.

179 M. L. Foresti, A. Vázquez and B. Boury, *Carbohydr. Polym.*, 2017, **157**, 447–467.

180 X. Kang, P. Sun, S. Kuga, C. Wang, Y. Zhao, M. Wu and Y. Huang, *ACS Sustainable Chem. Eng.*, 2017, **5**(3), 2529–2534.

181 Z. Wang, C. Crandall, V. L. Prautzsch, R. Sahadevan, T. J. Menkhaus and H. Fong, *ACS Appl. Mater. Interfaces*, 2017, **9**(4), 4272–4278.

182 N. N. Bui and J. R. McCutcheon, *Environ. Sci. Technol.*, 2013, **47**(3), 1761–1769.