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Boosting hydrogen generation by anodic oxidation of iodide over Ni–Co(OH)₂ nanosheet arrays†

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For overall water electrolysis, the hydrogen evolution reaction (HER) is severely limited by the sluggish kinetics of the anodic oxygen evolution reaction (OER). Therefore, replacing the OER with a more favorable anodic oxidation reaction with remarkable kinetics is of paramount significance, especially the one that can produce value-added chemicals. Moreover, time-saving and cost-effective strategies for the fabrication of electrodes are helpful for the wide application of electrolysis. Herein, thermodynamically more favorable iodide electrooxidation over Ni doped Co(OH)₂ nanosheet arrays (NSAs) in alkaline solution is presented as the alternative to the OER to boost the HER. And the active species are determined to be the reverse redox of the Co(IV)/Co(III) couple. Remarkably, a negative shift of voltage of 320 mV is observed at a current density of 10 mA cm⁻² after using iodide electrolysis to replace ordinary water splitting. The synthetic strategy and iodide oxidation in this work expand the application of Co-based materials in the field of energy-saving hydrogen production.

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Introduction

Electrocatalytic water splitting has been widely explored for the purpose of sustainable hydrogen generation.¹ However, the sluggish kinetics of the anodic oxygen evolution reaction (OER) always requires a high overpotential at high current density that severely inhibits the overall reaction rate.² Although a variety of outstanding electrocatalysts with attractive OER performance have been fabricated, the overpotential for the OER is still too high to match with the rate of the hydrogen evolution reaction (HER), thus increasing energy consumption.³ In this regard, replacing the OER with more favorable oxidation reactions at the anode has triggered a lot of interest, such as tetrahydroisoquinoline oxidation, the urea oxidation reaction (UOR), amine oxidation, and N₂ oxidation.^{4–15} Among them the UOR with a low equilibrium potential of 0.37 V has been regarded as a promising alternative to achieve energy-saving hydrogen generation and great process has been made.^{16–18} Unfortunately, the UOR also has sluggish kinetics due to its multi-electron process and requires highly active electrocatalysts to accelerate its reaction rate.¹⁹ Recently, nickel-based electrocatalysts have been demonstrated to have superior UOR activities with the oxidation of the Ni(III)/Ni(II) couple as active species.⁴ In this regard, the limitation of UOR performance has a strong relationship with the oxidation potential of Ni(II) species. Therefore, it is of great significance to explore other kinds of readily

oxidized chemicals with a fast response from the redox of other metal species, such as Co(IV)/Co(III) species that are not as efficient as Ni(III)/Ni(II) species for the UOR.^{20,21} Moreover, the oxidation products (CO₂ and N₂) of the UOR are of low value and the generated gas may also peel off the active materials from the substrate, lowering the electrocatalytic activity. Thus, it is significant but still challenging to achieve the generation of both hydrogen and value-added oxidation products with low energy consumption. Iodide with a lower oxidation potential than water should be easily oxidized compared to water during the electrocatalysis process.²² The oxidation product of iodide in alkaline solution is always oxyiodide that has been widely regarded as an efficient additive to salt for the compensation of iodine deficiency.^{23–25} Consequently, it is possible and meaningful to achieve energy-saving hydrogen generation through iodide electrolysis in an alkaline medium.

For the wide application of electrolysis for hydrogen production, another vital factor is to cut down the cost on the electrode fabrication process, which has long been ignored. Current strategies for the fabrication of electrodes can be classified into the following two categories: (1) direct coating of active materials onto current collectors,^{17,26–29} including glass carbon electrodes, metal foam, and carbon-based substrates; (2) *in situ* growth of active materials onto current collectors by different kinds of methods,^{30–34} such as hydrothermal/solvothermal methods, and electrodeposition. However, these two kinds of strategies also have some demerits. For example, polymer binders are inevitably involved in the coating strategy, which will inhibit the exposure of active sites. In addition, the massive gas evolution will also peel off the active materials from the current collector.³⁵ For the second strategy, a sophisticated

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synthesis procedure, complex precursor solution, or/and heat treatment are always required, which will increase the time and cost of electrode fabrication, although the attachment between the active materials and substrates can be greatly enhanced by using this kind of strategy.^{36–39} Thus, developing time-saving and cost-effective methods for the simple fabrication of electrodes towards electrolysis is of crucial significance, yet remains a big challenge.

Herein, based on the redox reaction between $\text{Co}(\text{NO}_3)_2$ solution and a Ni foam substrate, the electrode of Ni doped $\text{Co}(\text{OH})_2$ nanosheet arrays on Ni foam (Ni-Co(OH)₂ NSAs) can be simply fabricated at room temperature. Thanks to the easy oxidation feature of iodide, the anodic oxidation process can be efficiently triggered over the Ni-Co(OH)₂ NSA electrode. The required potential is merely 1.30 V (*versus* the reversible hydrogen electrode, *vs.* RHE) to drive a current density of 20 mA cm^{-2} , which is 30 and 240 mV lower than that of the UOR and OER, respectively. The similar performance of the Ni-free $\text{Co}(\text{OH})_2$ NSAs reveals that the reverse redox of the $\text{Co}(\text{IV})/\text{Co}(\text{III})$ couple is the active species for the IOR. Furthermore, the overall iodide electrolyzer assembled from the as-prepared Ni-Co(OH)₂ NSAs and a Ni-Mo electrode displays a low cell voltage of 1.34 V at 10 mA cm^{-2} and remarkable durability for over 12 h, further confirming the promising application of the IOR over Co-based materials for energy-saving hydrogen production.

Experimental

Synthetic procedures

Synthesis of Ni-Co(OH)₂ NSAs. Typically, $\text{Co}(\text{NO}_3)_2$ solution was obtained by dissolving 5 mmol of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ in 20 mL deionized (DI) water. Then the pre-treated Ni foam (sonicated with ethanol, 3 M HCl and DI water for 10 min, and washed with DI water for 10 times) with a size of $4 \times 2 \text{ cm}^2$ was dipped into the above solution, and maintained at room temperature for 5 h. The final product was washed with DI water and anhydrous ethanol, and then dried at room temperature for 12 h.

Synthesis of Ni(OH)₂ NSAs. The synthesis procedure is similar to that of the Ni-Co(OH)₂ NSAs but with replacing $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ with $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$.

Synthesis of Co(OH)₂ NSAs. Typically, ZIF-67 arrays were first fabricated according to the previous work.⁴⁰ Then the as-obtained ZIF-67 arrays were dipped into alcohol solution (20 mL of ethanol and 5 mL of DI water) and heated at 80 °C for 2 h. The final product was washed with DI water and anhydrous ethanol, and then dried at room temperature for 12 h.

Synthesis of the Ni-Mo electrode. A facile electrodeposition process was carried out similar to the reported work.⁴¹ Typically, the electrolyte was obtained by dissolving 10.5 mmol of $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$, 1 mmol of $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24} \cdot 4\text{H}_2\text{O}$, 21 mmol of sodium citrate tribasic dehydrate and 21 mmol of NaCl in 70 mL of DI water. The pH of the solution was adjusted to about 8 by adding ammonia. The electrochemical deposition of Ni-Mo was performed in a conventional two-electrode cell at a constant current density of 200 mA for 1000 s with pre-treated Ni foam ($1 \times 1 \text{ cm}^2$) as the working electrode and a Pt wire as the counter

electrode. Subsequently, the Ni foam was washed with deionized water to obtain the Ni-Mo electrode.

Characterization

XRD patterns were obtained through a Shimadzu XRD7000 powder X-ray diffractometer at a scan rate of $0.15^\circ \text{ min}^{-1}$ at room temperature. The morphologies were elucidated by FESEM carried out on a Hitachi S-4800 field emission scanning electron microscope. TEM tests were conducted on a HT-7700 X-MAX^{165T} with an acceleration voltage of 120 kV. A Thermo Scientific K-Alpha⁺ X-ray photoelectron spectrometer was utilized to collect XPS spectra. Inductively coupled plasma (ICP) analyses were conducted on a PE Nexion 2000.

Electrochemical measurements

Electrochemical measurements were conducted in a three-electrode cell through a 660E CH Instrument electrochemical workstation at room temperature with a carbon rod as the counter electrode and a saturated calomel electrode (SCE) as the reference electrode. The as-prepared electrode was directly utilized as the working electrode with a geometrical area of 1 cm^2 . The electrolyte for the OER, IOR and UOR was 1 M KOH, 1 M KOH with 0.33 M KI and 1 M KOH with 0.33 M urea, respectively. All potentials were calibrated according to $E_{(\text{RHE})} = E_{(\text{SCE})} + 0.059 \text{ pH} + 0.242$. The impedance data were collected through electrochemical impedance spectroscopy (EIS) measurements in the range of 0.01 to 100 000 Hz with 5 mV sinusoidal perturbations. Polarization curves were obtained at a scan rate of 5 mV s^{-1} . *iR* compensation was applied for all potentials. Tafel slopes were determined according to the following Tafel equation:

$$E = b \log j + a$$

where E is the potential (V), j is the current density (mA cm^{-2}), and b is the Tafel slope (mV dec^{-1}).

The electrochemical active surface area (ECSA) of the electrocatalysts was estimated according to the following equation:

$$\text{ECSA} = C_{\text{dl}}/C_s$$

where C_{dl} is the electric double layer capacitance; C_s is the specific surface capacitance, and a common C_s value of 0.04 was used in this work.⁴²

The Faraday efficiency of the IOR process was determined to be about 117.7%. The method to evaluate the Faraday efficiency of the IOR is described as follows and detailed in the Experimental section. The reason for the larger Faraday efficiency can be ascribed to the loss of AgI during the collection procedure.

Firstly, the IOR was conducted at a constant current of 50 mA for 2 h. Then, 35 mL of the electrolyte (total volume of the electrolyte at the anode is 35 mL) was obtained and 35 mL of Tollens was added (obtained by dissolving 14.0 g of AgNO_3 in 35 mL of ammonia) into the above electrolyte. Finally, the precipitate (AgI) was collected, washed with water and dried at 60 °C for 2 h. AgI is determined to be about 2.54 g. Under these



conditions the I^- anion can be precipitated after adding Tollens, while the IO_3^- anion cannot be precipitated. As a result, the Faraday efficiency can be calculated.

Results and discussion

The Ni–Co(OH)₂ NSAs were fabricated through a simple one-pot substrate oxidation strategy at room temperature, as schematically illustrated in Fig. 1. In the synthetic process, the hydrolysis of $Co(NO_3)_2$ will release H^+ ions to build an acidic condition, which can promote the oxidation ability of NO_3^- ions. Then Ni foam can be oxidized by NO_3^- ions with the help of H^+ ions to deliver Ni^{2+} species. Finally, with the consumption of H^+ ions in the solution, pH near the Ni foam surface will increase thus generating Ni–Co(OH)₂ NSAs. The crystallographic structure of the as-obtained electrocatalyst was first elucidated by the powder X-ray diffraction (XRD) test. As shown in Fig. S1,† except the three strong peaks of Ni foam, all of the rest of the peaks can be well indexed to $Co(OH)_2$ (JCPDS card No. 51-1731). The morphology of the as-obtained electrocatalyst was examined by field emission scanning electron microscopy (FESEM). Panoramic FESEM images in Fig. 2A and B illustrate that the Ni foam surface is covered by a layer of interconnected nanosheets with vertical alignment. The distance between the adjacent nanosheets is several hundred nanometers, which is favorable for the penetration of electrolyte and transfer of reactants. In sharp contrast, only a flat surface is observed for a pristine Ni foam substrate (Fig. S2†). The nanosheet morphology of the Ni–Co(OH)₂ NSAs is further confirmed by

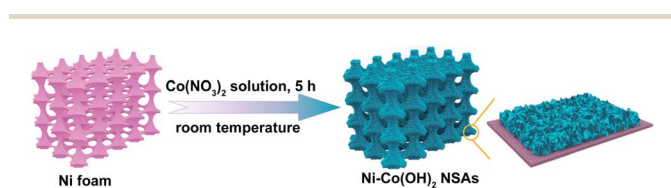


Fig. 1 Schematic illustration of the fabrication of Ni–Co(OH)₂ NSAs.

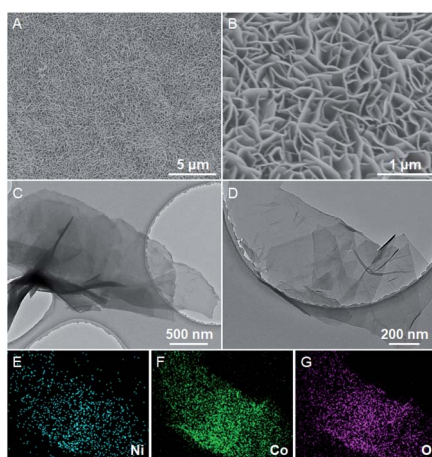


Fig. 2 SEM (A and B) and TEM (C and D) images of Ni–Co(OH)₂ NSAs. Mapping images of different elements for the nanosheets of Ni (E), Co (F) and O (G).

transition electron microscopy (TEM) images (Fig. 2C and D), in which a smooth surface with lateral size of hundreds of nanometers can be observed. Further, the spatial dispersion distribution of the Ni, Co and O elements in the nanosheet was elucidated by elemental mapping examination (Fig. 2E–G). As can be seen, the three elements homogeneously disperse in the whole nanosheet, indicating the uniform incorporation of Ni species. Raman scattering spectroscopy was conducted to further analyze the chemical structure of Ni–Co(OH)₂ NSAs. Two main peaks can be observed at about 470 and 531 cm^{-1} in the Raman spectrum (Fig. S3†), corresponding to the characteristic peaks for Ni–O and Co–O vibrational modes.⁴³ All of the above results imply the successful fabrication of Ni–Co(OH)₂ NSAs with a homogeneous Ni distribution.

X-ray photoelectron spectroscopy (XPS) analyses were performed to illustrate the surface chemical composition and oxidation state of Ni–Co(OH)₂ NSAs. As shown in Fig. S4A,† the XPS spectrum of Ni–Co(OH)₂ NSAs demonstrates the coexistence of Ni, Co, and O elements in the electrocatalyst. The spin-orbit doublet in the high-resolution spectrum of Co 2p (Fig. S4B†) centered at 781.14 and 796.78 eV can be indexed to the Co(II) species with two satellite peaks at 785.67 and 802.85 eV. And the differential binding energy of 15.64 eV between these two peaks further prove the oxidation state of +2 for Co species in the electrocatalyst, consistent with previous literature.⁴⁴ Similarly, there are also a spin-orbit doublet and two satellite peaks in the high-resolution spectrum of Ni 2p (Fig. S4C†). The peaks at 855.87 and 873.46 eV correspond to the characteristic peaks of Ni(II) species in the electrocatalyst.⁴⁵ And the peaks assigned to satellite peaks are observed at 861.54 and 879.26 eV. The high-resolution spectrum of O 1s uncovers the existence of three types of oxygen species, including metal-oxygen bonding at 530.70 eV, OH^- species at 531.39 eV, and absorbed oxygen at 532.38 eV.⁴⁶ The above results further confirm the successful incorporation of Ni species into $Co(OH)_2$.

The electrocatalytic oxidation reaction of iodide (IOR) was performed in 1 M KOH solution and 0.33 M KI solution. Firstly, the optimal reaction time was determined through evaluating the IOR activity of the electrocatalysts synthesized with different times. The polarization curves clearly present that the highest activity can be achieved with a reaction time of 5 h (Fig. S5†). As can be seen in Fig. 3A, the potential to drive a current density of 20 $mA\ cm^{-2}$ is determined to be 1.30 V (vs. RHE). To further demonstrate the superiority of the IOR on lowering the potential at the same current density, UOR (0.33 M urea) and OER measurements were also carried out. The required potentials at 20 $mA\ cm^{-2}$ are as high as 1.33 and 1.54 V (vs. RHE) for the UOR and OER, respectively. Similar to the UOR process, the key to trigger the IOR process is the generation of high valence state metal species. As presented in the polarization curve of the OER (Fig. 3A), three obvious oxidation peaks can be found at about 1.09, 1.31 and 1.39 V (vs. RHE) that refer to the oxidation process of $Co(II)/Co(III)$, $Co(III)/Co(IV)$, and $Ni(II)/Ni(III)$,^{47,48} respectively. After adding KI into electrolyte, only the oxidation peak of $Co(II)/Co(III)$ can be observed in the polarization curve, implying that the oxidation of KI occurred at the potential where high-valence



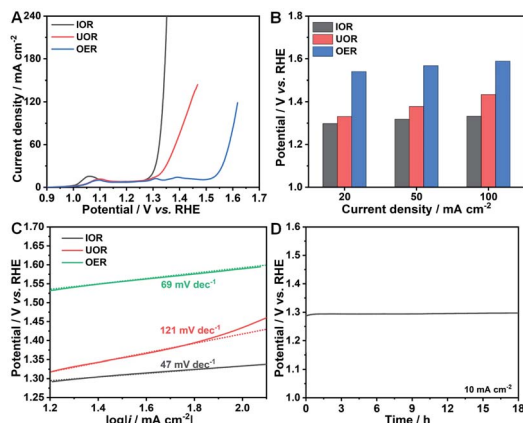


Fig. 3 (A) Polarization curves of the IOR, UOR and OER over Ni-Co(OH)₂ NSAs. (B) Comparison of potentials required to drive different current densities of the IOR, UOR and OER over Ni-Co(OH)₂ NSAs. (C) Tafel plots of electrooxidation reactions from (A). (D) Chronopotentiometry curve of the IOR over Ni-Co(OH)₂ NSAs.

metal species (Co(IV) and Ni(III)) were generated. The faster increase of the anodic current density of the IOR than that of the UOR indicates a more favorable response of the IOR process than the UOR process driven from the redox of the Co(III)/Co(IV) couple.

For the IOR, the potentials to reach current densities of 50 and 100 mA cm⁻² are determined to be 1.32 and 1.33 V (vs. RHE), whereas they are much higher for UOR and OER processes (Fig. 3B). The highest activity for the IOR among the three electrocatalytic oxidation reactions well illustrate its advantage as a promising alternative to the OER to lower the energy consumption on hydrogen generation. The Tafel slopes of these three electrocatalytic oxidation reactions were calculated to evaluate their reaction kinetics. The Tafel slope for the IOR is only 47 mV dec⁻¹, while it is 121 and 69 mV dec⁻¹ for the UOR and OER, respectively (Fig. 3C). The smallest Tafel slope of the IOR points to its more favorable catalytic kinetics than the UOR and OER. Electrocatalytic durability, one of the most crucial evaluation criteria, is important to be measured. Accordingly, chronopotentiometry measurements at a constant current density of 10 mA cm⁻² were carried out. Remarkably, the IOR process is quite durable with merely negligible changes in the potential over 18 h (Fig. 3D). And it also can smoothly work at 100 mA cm⁻² for 5 h (Fig. S6A†). Moreover, the recorded multi-current curve (Fig. S6B†) displays the rapid response to the current change of Ni-Co(OH)₂ NSAs towards the IOR. Once the current is changed, the required potential can immediately level off and be well preserved for 1000 s, suggesting the superior mass transfer properties and robust structure of Ni-Co(OH)₂ NSAs towards the IOR. The ultrahigh activity and stability of the IOR over Ni-Co(OH)₂ NSAs suggest that the IOR is a promising oxidation reaction to replace the OER for energy-saving hydrogen generation. Moreover, the application of Co-based electrocatalysts in the field of energy-saving hydrogen generation can be expanded at the same time through the IOR. The morphology of the Ni-Co(OH)₂ NSAs after IOR durability

measurements was measured. As shown in Fig. S7A-C,† the nanosheet arrays and microstructures are well preserved after the IOR durability test, suggesting the robust structure of Ni-Co(OH)₂ NSAs. ICP analyses reveal that the atomic ratios of Co and Ni elements before and after the long-term reaction are evaluated to be 6.80 and 7.17, respectively. These results imply that some Co will be dissolved into the electrolyte during the IOR process, though the structure of Ni-Co(OH)₂ NSAs can be well retained. In addition, XRD and XPS measurements were further conducted to elucidate the changes on Ni-Co(OH)₂ NSAs after performing the IOR. As presented in the XRD pattern (Fig. S7D†), the peaks indexed to Co(OH)₂ disappear, while some new weak peaks corresponding to CoOOH can be observed. XPS results also reveal the changes of the chemical oxidation state of elements. Although Co, Ni and O elements still exist in the electrocatalysts (Fig. S4A†), the oxidation states for these elements are obviously different, especially the Co and O. Two new peaks centered at 779.57 and 794.61 eV with an energy difference of 15.04 eV in the Co 2p spectrum after the IOR can be indexed to Co(III) species (Fig. S4C†).⁴⁹ For the O 1s spectrum, the peaks at 529.11 and 530.90 eV can be attributed to Co-O and Co-OH bonds, respectively (Fig. S4D†).⁵⁰ And the typical peak for absorption of oxygen is also observed at a binding energy of 532.38 eV. The binding energies of Ni species only exhibit a slightly negative shift of about 0.23 and 0.25 eV for Ni 2p_{3/2} and Ni 2p_{1/2}, respectively, with the chemical oxidation state unchanged. The phase transformation and chemical oxidation state change after the IOR process implies that the redox of Co(III)/Co(IV) and Ni(II)/Ni(III) couples is the key to the IOR, coincident with the electrochemical measurements. In the IOR process, Co(II) species are firstly oxidized to Co(III) species and Co(IV) at a high potential. Similarly, the Ni(II) species are also oxidized to Ni(III) at a high potential. Then, the I⁻ ions can immediately transfer to IO³⁻ ions once they attach the newly formed Co(IV) and Ni(III) species that will be reduced to Co(III) and Ni(II) species at the same time. As a result, the electrocatalyst after IOR tests is determined to be CoOOH with doped Ni(II).

To have a clear understanding of the crucial importance of Co species in lowering the IOR potential, Ni(OH)₂ on Ni foam was also fabricated. The XRD pattern and FESEM images (Fig. S1 and S8†) well confirm the successful formation of Ni(OH)₂ with the morphology of nanosheets on Ni foam. The activities towards the IOR of the as-obtained NSAs was recorded by the polarization curve. As shown in Fig. 4A, the IOR performance of Ni-Co(OH)₂ NSAs far more outperforms that of Ni(OH)₂ NSAs. The potential to reach a current density of 20 mA cm⁻² is 1.34 V (vs. RHE) for Ni(OH)₂ NSAs, which is almost 40 mV higher than that of Ni-Co(OH)₂ NSAs. The IOR activity of the pristine Ni foam substrate was also evaluated and the required potential is 1.43 V (vs. RHE) at 20 mA cm⁻². The Tafel plots of the recorded polarization curves are presented in Fig. 4B. Remarkably, the Tafel slopes of Ni-Co(OH)₂ NSAs and Ni(OH)₂ NSAs are 47 and 49 mV dec⁻¹, while it is as high as 74 mV dec⁻¹ for pristine Ni foam. The similar Tafel slopes of both NSAs suggest that these two electrocatalysts have similar reaction kinetics for the IOR. The ECSA of the Ni-Co(OH)₂ NSAs and



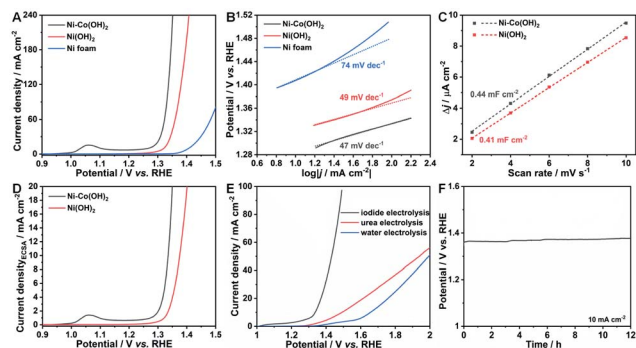


Fig. 4 (A) Polarization curves of the IOR over Ni–Co(OH)₂ NSAs, Ni(OH)₂ NSAs and Ni foam. (B) The corresponding Tafel plots from (A). (C) Difference in current density plotted against the scan rate displaying the value of double-layer capacitances. (D) ECSA normalized polarization curves of Ni–Co(OH)₂ NSAs, and Ni(OH)₂ NSAs from (A). (E) Polarization curves of iodide electrolysis, urea electrolysis, and water electrolysis over the Ni–Co(OH)₂ NSAs||Ni–Mo electrode pair. (F) Chronopotentiometry curve of iodide electrolysis at a current density of 10 mA cm⁻².

Ni(OH)₂ NSAs was further measured to uncover their intrinsic activity towards the IOR. Since the ECSA is linearly proportional to the double-layer capacitances (C_{dl}), we first determined C_{dl} through cyclic voltammetry (CV) measurements at scan rates from 10 to 50 mV s⁻¹. According to CV curves in Fig. S9,† the C_{dl} of Ni–Co(OH)₂ NSAs and Ni(OH)₂ NSAs was calculated to be 0.44 and 0.41 mF cm⁻² (Fig. 4C), corresponding to the ECSA of 11 and 10.25 cm², respectively. This result implies that the superior IOR performance for Ni–Co(OH)₂ NSAs is independent of the ECSA, since both the electrocatalysts exhibit similar ECSAs with quite different current density at the same potential (Fig. 4D). The current density normalized to the ECSA of Ni–Co(OH)₂ NSAs is higher than that of Ni(OH)₂ NSAs, suggesting the remarkable intrinsic activity of Ni–Co(OH)₂ NSAs towards the IOR, which can be attributed to its favorable response driven by the redox of Co(III)/Co(IV) species.

Furthermore, the overall electrolysis of iodide, urea and water was carried out in a two electrode cell by using the as-obtained Ni–Co(OH)₂ NSAs as the anode and the Ni–Mo electrode as the cathode. The Ni–Mo electrode was fabricated by a facile electrodeposition method according to the literature,⁴¹ and its morphology was characterized by FESEM (Fig. S10A and B†). The utilization of the Ni–Mo electrode as the cathode in this system can be ascribed to its low cost, convenient fabrication, and extremely superior HER activity with a low overpotential of 43 mV at 10 mA cm⁻² (Fig. S10C†). As expected, the overall iodide electrolysis displays prominent activity by delivering a current density of 10 mA cm⁻² at only 1.34 V, while it is as high as 1.49 and 1.66 V for urea and water electrolysis, respectively (Fig. 4E). Further, overall iodide electrolysis can operate smoothly for at least 12 h, confirmed by the chronopotentiometry test (Fig. 4F). These remarkable features imply that Ni–Co(OH)₂ NSAs are an excellent electrocatalyst for the IOR and the IOR process can be one of the promising alternatives to the OER process to boost hydrogen generation efficiency with low energy consumption.

To exclude the effect of Ni ions on the IOR performance, a pure Co(OH)₂-based electrocatalyst on Ni foam without Ni incorporation was synthesized with ZIF-67 as a precursor similar to the previous work of the Lou group.⁵¹ FESEM images (Fig. S11†) and XRD patterns (Fig. S12†) reveal the successful fabrication of ZIF-67 and Co(OH)₂ on Ni foam with a triangular and nanosheet shape (denoted as Co(OH)₂ NSAs), respectively. The IOR, UOR and OER performances of Co(OH)₂ NSAs were also evaluated. As seen in polarization curves presented in Fig. 5A, the electrocatalytic performance for the IOR is significantly remarkable with a low potential of 1.31 V (vs. RHE) to drive a current density of 20 mA cm⁻². However, the UOR and OER activities over Co(OH)₂ NSAs are quite sluggish. The required potential at 20 mA cm⁻² for the UOR is as high as 1.40 V (vs. RHE), and it is even higher than 1.5 V (vs. RHE) for the OER. The IOR performance over the as-prepared Co-based electrocatalysts is superior to that of other kinds of anodic oxidation reactions (Fig. 5B and Table S1†), such as the UOR over CoFeCr LDH/NF,⁵² S–MnO₂,¹⁷ CoMn/CoMn₂O₄,⁵³ and NF/NiMoO–Ar,¹⁶ and 5-hydroxymethylfurfural (HMF) oxidation over Ni_xB,²⁷ Ni₂P NPA/NF,⁵⁴ NiCo₂O₄,⁵⁵ and Ni₃S₂/NF.⁵⁶ Tafel slopes are calculated to be 52, 115 and 139 mV dec⁻¹ for the IOR, UOR and OER, respectively, indicating more favorable catalytic kinetics of the IOR than the UOR and OER (Fig. 5C). The negligible change of potential over 18 h in chronopotentiometry measurements conducted at a constant current density of 20 mA cm⁻² (Fig. 5D) suggests the prominent stability of the IOR over Co(OH)₂ NSAs. All of the above results further imply that the Co-based electrocatalysts exhibit more efficient response to the IOR than to UOR and OER. And the Co(III)/Co(IV) couple is of key importance to lower the potential for iodide electrooxidation. It is worth noting that the application of Co-based electrocatalysts for energy-saving hydrogen generation can be expanded by the IOR process.

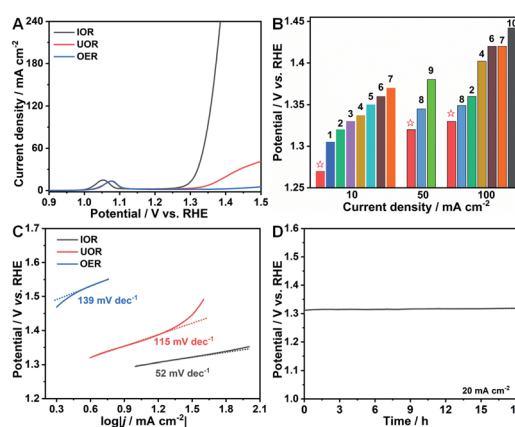


Fig. 5 (A) Polarization curves of the IOR, OER, and UOR over Co(OH)₂ NSAs. (B) Comparison of potentials at 10, 50, and 100 mA cm⁻² of Ni–Co(OH)₂ NSAs in this work with those of other representative electrocatalysts in references. (1): CoFeCr LDH/NF,⁵² (2): CoMn/CoMn₂O₄,⁵³ (3): S–MnO₂,¹⁷ (4): C-350,⁶ (5): Ni₃N NA/CC,⁷ (6): FQD/CoNi-LDH/NF,⁸ (7): NF/NiMoO–Ar,¹⁶ (8): NiIr-MOF/NF,⁹ (9): MoS₂/CoS/Co_{0.85}Se HNT,³⁴ and (10): MS–Ni₂P/Ni_{0.96}S/NF.¹⁰ (C) Corresponding Tafel plots from (A). (D) Chronopotentiometry curve of the IOR over Co(OH)₂ NSAs at a current density of 20 mA cm⁻².



Conclusions

In summary, the highly active electrode of Ni-Co(OH)₂ NSAs was successfully fabricated at room temperature by simply immersing Ni foam in Co(NO₃)₂ solution. And we also demonstrated that the electrooxidation of iodide can efficiently take place over Ni-Co(OH)₂ NSAs with favorable kinetics and high durability, in which Co(IV)/Co(III) serves as the primary active species to accelerate the reaction rate. Thanks to the ready oxidation feature of iodide, integrating with the Ni-Mo electrode, overall iodide electrolysis exhibits an ultralow cell voltage of 1.34 V at 10 mA cm⁻², which is almost 320 mV lower than that of water electrolysis. The outstanding stability also illustrates that the as-prepared Ni-Co(OH)₂ NSAs are a promising electrocatalyst towards the IOR in an alkaline medium. We believe that this work can open a new avenue for the application of Co-based materials as anodic oxidation reaction electrocatalysts for energy-saving hydrogen generation with value-added oxidation products generated.

Conflicts of interest

There are no conflicts to declare.

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References

- Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov and T. F. Jaramillo, *Science*, 2017, **355**, eaad4998.
- B. You and Y. Sun, *Acc. Chem. Res.*, 2018, **51**, 1571–1580.
- J.-B. Tan and G.-R. Li, *J. Mater. Chem. A*, 2020, **8**, 14326–14355.
- B. Zhu, Z. Liang and R. Zou, *Small*, 2020, **16**, 1906133.
- L. Chen and J. Shi, *J. Mater. Chem. A*, 2018, **6**, 13538–13548.
- S. Hu, S. Wang, C. Feng, H. Wu, J. Zhang and H. Mei, *ACS Sustainable Chem. Eng.*, 2020, **8**, 7414–7422.
- Q. Liu, L. Xie, F. Qu, Z. Liu, G. Du, A. M. Asiri and X. Sun, *Inorg. Chem. Front.*, 2017, **4**, 1120–1124.
- Y. Feng, X. Wang, J. Huang, P. Dong, J. Ji, J. Li, L. Cao, L. Feng, P. Jin and C. Wang, *Chem. Eng. J.*, 2020, **390**, 124525.
- Y. Xu, X. Chai, T. Ren, S. Yu, H. Yu, Z. Wang, X. Li, L. Wang and H. Wang, *Chem. Commun.*, 2020, **56**, 2151–2154.
- M. He, C. Feng, T. Liao, S. Hu, H. Wu and Z. Sun, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2225–2233.
- B. You, G. Han and Y. Sun, *Chem. Commun.*, 2018, **54**, 5943–5955.
- C. Huang, Y. Huang, C. Liu, Y. Yu and B. Zhang, *Angew. Chem., Int. Ed.*, 2019, **58**, 12014–12017.
- Y. Huang, X. Chong, C. Liu, Y. Liang and B. Zhang, *Angew. Chem., Int. Ed.*, 2018, **57**, 13163–13166.
- X. Chong, C. Liu, Y. Huang, C. Huang and B. Zhang, *Natl. Sci. Rev.*, 2019, **7**, 285–295.
- Y. Wang, Y. Yu, R. Jia, C. Zhang and B. Zhang, *Natl. Sci. Rev.*, 2019, **6**, 730–738.
- Z. Yu, C. Lang, M. Gao, Y. Chen, Q. Fu, Y. Duan and S. Yu, *Energy Environ. Sci.*, 2018, **11**, 1890–1897.
- S. Chen, J. Duan, A. Vasileff and S. Z. Qiao, *Angew. Chem., Int. Ed.*, 2016, **55**, 3804–3808.
- D. Liu, T. Liu, L. Zhang, F. Qu, G. Du, A. M. Asiri and X. Sun, *J. Mater. Chem. A*, 2017, **5**, 3208–3213.
- X. Hu, J. Zhu, J. Li and Q. Wu, *ChemElectroChem*, 2020, **7**, 3211–3228.
- L. Sha, K. Ye, G. Wang, J. Shao, K. Zhu, K. Cheng, J. Yan, G. Wang and D. Cao, *J. Power Sources*, 2019, **412**, 265–271.
- N. Senthilkumar, G. Gnana kumar and A. Manthiram, *Adv. Energy Mater.*, 2018, **8**, 1702207.
- A. J. Bard and L. R. Faulkner, *ELECTROCHEMICAL METHODS-Fundamentals and Applications*, JOHN WILEY & SONS, INC, 2001.
- G. Arroyave, O. Pineda and N. S. Scrimshaw, *Bull. W. H. O.*, 1956, **14**, 183–185.
- X. Li, X. Cao, J. Li, J. Xu, W. Ma, H. Wang, J. Wang and Y. Zhang, *J. Trace Elem. Med. Biol.*, 2020, **62**, 126575.
- H. de Santana and M. L. A. Temperini, *J. Electroanal. Chem.*, 1991, **316**, 93–105.
- X. Zhu, X. Dou, J. Dai, X. An, Y. Guo, L. Zhang, S. Tao, J. Zhao, W. Chu, X. C. Zeng, C. Wu and Y. Xie, *Angew. Chem., Int. Ed.*, 2016, **55**, 12465–12469.
- S. Barwe, J. Weidner, S. Cychy, D. M. Morales, S. Dieckhöfer, D. Hiltrop, J. Masa, M. Muhler and W. Schuhmann, *Angew. Chem., Int. Ed.*, 2018, **57**, 11460–11464.
- J. Wu, D. Wang, S. Wan, H. Liu, C. Wang and X. Wang, *Small*, 2020, **16**, 1900550.
- E. Hu, Y. Feng, J. Nai, D. Zhao, Y. Hu and X. W. Lou, *Energy Environ. Sci.*, 2018, **11**, 872–880.
- Y. Tong, P. Chen, M. Zhang, T. Zhou, L. Zhang, W. Chu, C. Wu and Y. Xie, *ACS Catal.*, 2018, **8**, 1–7.
- T. Liu, D. Liu, F. Qu, D. Wang, L. Zhang, R. Ge, S. Hao, Y. Ma, G. Du, A. M. Asiri, L. Chen and X. Sun, *Adv. Energy Mater.*, 2017, **7**, 1700020.
- M. Ranjani, N. Senthilkumar, G. Gnana kumar and A. Manthiram, *J. Mater. Chem. A*, 2018, **6**, 23019–23027.
- J. Xie, W. Liu, F. Lei, X. Zhang, H. Qu, L. Gao, P. Hao, B. Tang and Y. Xie, *Chem.-Eur. J.*, 2018, **24**, 18408–18412.
- Y. Sun, S. Wang, J. Ning, Z. Zhang, Y. Zhong and Y. Hu, *Nanoscale*, 2020, **12**, 991–1001.
- K. Zeng and D. Zhang, *Prog. Energy Combust. Sci.*, 2010, **36**, 307–326.
- X. Lu and C. Zhao, *Nat. Commun.*, 2015, **6**, 6616.
- Y. Li, P. Hasin and Y. Wu, *Adv. Mater.*, 2010, **22**, 1926–1929.
- L. Sha, K. Ye, J. Yin, K. Zhu, K. Cheng, J. Yan, G. Wang and D. Cao, *Chem. Eng. J.*, 2020, **381**, 122603.
- E. Hu, J. Ning, D. Zhao, C. Xu, Y. Lin, Y. Zhong, Z. Zhang, Y. Wang and Y. Hu, *Small*, 2018, **14**, 1704233.
- C. Guan, W. Zhao, Y. Hu, Z. Lai, X. Li, S. Sun, H. Zhang, A. K. Cheetham and J. Wang, *Nanoscale Horiz.*, 2017, **2**, 99–105.
- M. Xia, T. Lei, N. Lv and N. Li, *Int. J. Hydrogen Energy*, 2014, **39**, 4794–4802.



- 42 C. C. L. McCrory, S. Jung, J. C. Peters and T. F. Jaramillo, *J. Am. Chem. Soc.*, 2013, **135**, 16977–16987.
- 43 X. Cai, X. Shen, L. Ma, Z. Ji, C. Xu and A. Yuan, *Chem. Eng. J.*, 2015, **268**, 251–259.
- 44 Y. Luo, X. Li, X. Cai, X. Zou, F. Kang, H.-M. Cheng and B. Liu, *ACS Nano*, 2018, **12**, 4565–4573.
- 45 F. Chen, C. Chen, Q. Hu, B. Xiang, T. Song, X. Zou, W. Li, B. Xiong and M. Deng, *Chem. Eng. J.*, 2020, **401**, 126145.
- 46 H. Liang, J. Lin, H. Jia, S. Chen, J. Qi, J. Cao, T. Lin, W. Fei and J. Feng, *J. Mater. Chem. A*, 2018, **6**, 15040–15046.
- 47 M. Yang, H. Cheng, Y. Gu, Z. Sun, J. Hu, L. Cao, F. Lv, M. Li, W. Wang, Z. Wang, S. Wu, H. Liu and Z. Lu, *Nano Res.*, 2015, **8**, 2744–2754.
- 48 L. Peng, L. Lv, H. Wan, Y. Ruan, X. Ji, J. Liu, L. Miao, C. Wang and J. Jiang, *Materials Today Energy*, 2017, **4**, 122–131.
- 49 J. Yang, H. Liu, W. N. Martens and R. L. Frost, *J. Phys. Chem. C*, 2010, **114**, 111–119.
- 50 Z. Wang, L. Wang, S. Liu, G. Li and X.-P. Gao, *Adv. Funct. Mater.*, 2019, **29**, 1901051.
- 51 H. Hu, B. Y. Guan and X. W. Lou, *Chem*, 2016, **1**, 102–113.
- 52 Z. Wang, W. Liu, Y. Hu, M. Guan, L. Xu, H. Li, J. Bao and H. Li, *Appl. Catal., B*, 2020, **272**, 118959.
- 53 C. Wang, H. Lu, Z. Mao, C. Yan, G. Shen and X. Wang, *Adv. Funct. Mater.*, 2020, **30**, 2000556.
- 54 B. You, N. Jiang, X. Liu and Y. Sun, *Angew. Chem., Int. Ed.*, 2016, **55**, 9913–9917.
- 55 M. J. Kang, H. Park, J. Jegal, S. Y. Hwang, Y. S. Kang and H. G. Cha, *Appl. Catal., B*, 2019, **242**, 85–91.
- 56 B. You, X. Liu, N. Jiang and Y. Sun, *J. Am. Chem. Soc.*, 2016, **138**, 13639–13646.

