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Ultrafast probe at the interfaces of solar energy conversion materials

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Abstract

Ultrafast carrier dynamics at interfaces plays a major role in governing solar energy conversion efficiency in solar conversion systems, especially in solar cells and photoelectrochemical devices. In this perspective, we discuss recent advances in ultrafast spectroscopic probes to understand the dynamics at such interfaces. Transient reflectance (TR) enables monitoring of the surface carriers that are within ~ 10 nm of the surface or interface of interest. Crucial parameters such as surface recombination velocity and carrier diffusion can be determined. When a strong surface or interfacial field is present the reflectance spectrum will exhibit Franz-Kelydesh oscillations. Monitoring the transient formation or decay of such oscillations informs upon the interfacial field and is referred to as Transient Photoreflectance (TPR). Such fields are helpful in designing systems in order to move carriers away from surfaces where photocorrision can occur and towards catalytic surfaces. Finally, we discuss transient infrared attenuate total reflection (TATR) to probe the ultrafast reaction intermediates in a photo-driven chemical reaction.

Introduction

Solar energy represents one of the most promising renewable energy sources. It can be converted to many forms of energy such as heat, electricity, and chemical fuels. ¹⁻³ The conversion of solar energy to electricity and fuels are probably the most valuable to human society since they form the infrastructure of modern society. ²⁻³ The conversion of solar energy to electricity requires the use of solar cells (photovoltaics). In this type of device, sun light is absorbed by a semiconductor that has a positive and negative electrode contact. Absorption of light promotes electrons from the valence to the conduction band leaving behind a hole in the valence band. Photogenerated electrons and holes are separated

within the device at specific interfaces that accept only electrons or holes and then transported into an external circuit where they can do work prior to recombining, which completes the solar to electricity conversion. Currently, Si solar cells dominate the commercial market, but other systems such as GaAs, CdTe, lead-halide perovskites and quantum dots are emerging solar cell technologies. Solar fuel generation represents another area of solar energy research. The direct conversion of solar energy to a chemical fuel requires a photocatalytic system whereby light is absorbed by a photo active material generating energetic electrons (reducing equivalents) and holes (oxidizing equivalents). In a typical design the photogenerated electrons and holes transfer to a photocatalysts where the specific reduction and oxidation reactions that generate the fuel can occur. Sometimes the photo active material can itself serve as the photocatalysts. The most common solar fuel generation processes include water splitting and CO₂ reduction on semiconducting light absorbing photoelectrodes such as Si, GaAs, GaInP₂ or metal oxides. The chemical fuel produced from these photo-driven reactions include molecular hydrogen, oxygen, carbon monoxide or simple alcohols.

The development of novel materials for solar energy conversion requires knowledge from many areas of chemistry and physics, including design, synthesis, characterizations and most importantly, mechanistic understanding so as to direct the solar energy into desired products with little loss. The understanding of charge-carrier dynamics and the resulting chemical intermediates that form at the catalyst provide key insights in designing and synthesizing new materials, architectures, and systems. ¹⁰⁻¹¹ More importantly, in solar energy conversion systems, charge-carrier dynamics at or near interfaces, such as, surface-carrier recombination, surface-charge separation, interfacial electric field kinetics, charge transfer and the formation of photochemical reaction intermediates play a key role in determining the final solar energy conversion efficiencies ¹²⁻¹⁴ as well as the stability of the semiconductor absorber layers. The primary events of the above processes generally occur within the ultrafast time window, ranging from a few picoseconds (ps) to several nanoseconds (ns). ⁴ Thus, to directly monitor interfacial carrier dynamics, ultrafast probes are needed. Recently, with the development in ultrafast laser sources, non-contact spectroscopic techniques have been introduced to study such phenomenon at the

interfaces of interest. In this perspective, we will introduce and focus on three important types of ultrafast techniques to understand interfacial carrier dynamics. We will first discuss the studies of surface charge carrier recombination dynamics with transient reflection (TR) spectroscopy. Then we will introduce the transient photoreflectance (TPR) technique to understand ultrafast interfacial field dynamics and charge separation. Last but not least, we will introduce ultrafast techniques to identify early reaction intermediates on surfaces in solar fuel systems. Figure 1 highlights the ultrafast processes of interest in solar cell and solar fuel applications.

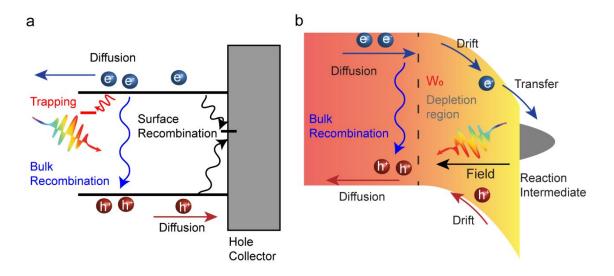


Figure 1. Ultrafast surface carrier processes in a. representative solar cells and b. solar fuel systems. The semiconductor illustrated here is p-type which has downward band bending. In **a** the solar cell conversion efficiency is given by a competition between carrier transport to charge-separating interfaces and any other recombination processes including radiative, or non-radiative process at point defects, grain boundaries, and surfaces or interfaces. In **b.** built-in fields at interfaces can be used to separate carriers (electrons are driven to the surface and holes away from the surface). The primary process is the charge transfer from the photoactive absorber to the surface where the reagents are in contact.

Surface charge carrier dynamics in solar cells probed by transient reflection (TR) spectroscopy

Surface carrier dynamics in solar cells has a huge impact on the performance because it can drastically reduce carrier-lifetimes and thus lower the quasi-fermi levels which governs the attainable short-circuit current and open-circuit voltage. ¹⁵⁻¹⁶ In high efficiency solar energy conversion applications, surface modification (passivation) becomes a necessary

component in order to reduce surface recombination.¹⁷ The knowledge of surface carrier dynamics can provide guidance for different surface modification strategies which can optimize solar cell performance. To probe the surface carrier dynamics, several surface sensitive probes such as time-resolved photoluminescence (TRPL)^{11, 18}, time-resolved terahertz spectroscopy (TRTS)¹⁹ and transient reflection spectroscopy (TR)²⁰⁻²¹ have been developed. All these techniques are important to the solar cell research. In the scope of this perspective, we will introduce the transient reflection spectroscopy as an ultrafast probe for surface carrier dynamics in solar cell materials and architectures.

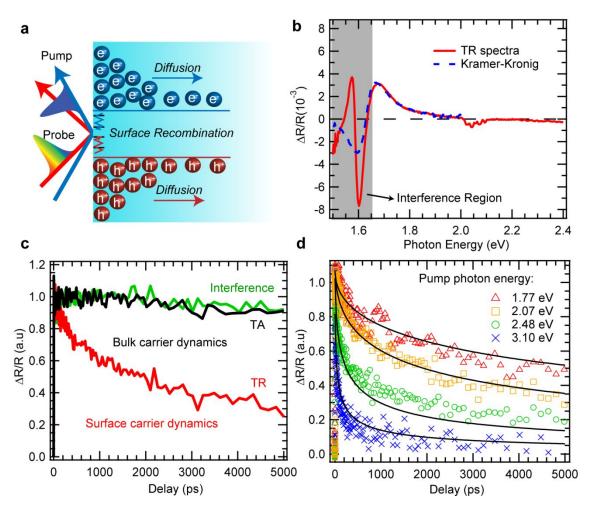


Figure 2. Surface carriers probed by ultrafast transient reflection a. Schematic illustration of transient reflection at solar cell surfaces to probe surface carrier dynamics. Ultrafast pump pulse generates a distribution of charge carriers (electrons and holes), then after time delay, an ultrafast broadband probe pulse monitor carrier diffusion and surface recombination of charge carriers. **b.** Representative transient reflection spectra for MAPbI₃ thin film solar cells at 2 ps delay. Kramer-Kronig relation can be used to extract the change

in absorption coefficient as shown in the blue dashed line. In thin films, interference pattern can be observed in the transient spectra resulting from the front and back surface reflection. $\bf c$. Thus, we can monitor both the transient kinetics of carriers in the bulk, whose contributions are recorded in the interference region, and transient kinetic of carriers near the surface, whose contribution is outside the interference region. For thin films, the transient kinetics in the interference region (green-trace) matches the bulk carrier dynamics (black traces) measured by transient absorption (TA) while the kinetics outside interference region represents surface carrier dynamics. (TR, red-trace) $\bf d$. The normalized TR kinetics recorded near the bandgap of MAPbI₃ outside the interference region for four different pump energies. The black dashed lines represent the global fitting based on the carrier diffusion and surface recombination model. Reproduced with permission from ref 21. Copyright 2017 Nature Publishing Group.

Transient reflection spectroscopy (TR) is a pump-probe based technique that is sensitive to photogenerated carriers (also excitons) in semiconductors. (See Figure 2a for schematic illustration of a typical TR setup) In principle, it involves excitation of the sample with a short (from a few hundred femtoseconds to a few picoseconds) laser pump pulse, followed by a weak time-delayed broadband white light continuum (WLC) probing pulse that spans the bandgap of the semiconductor surface of interest. In a typical ultrafast system, the WLC probe light can range from 350 nm to 1600 nm, spanning all the semiconductors relevant for solar energy conversion. The WLC reflects off the sample surfaces and the measured quantity is the pump induced change in reflectance ($\Delta R/R$). Typically, near the bandedge of semiconductors the real part of the complex refractive index dominates over the imaginary part $(n \gg k)$. Thus, we only consider how the photoinduced carriers modulate the real part of the refractive index (n). Due to the reflection geometry, the probe light only penetrates to a very shallow depth into the material, with a penetration depth on the order to 10 - 50nm depending on the refractive index. (since $k \ll n$, $d_{probe} \cong \lambda/4\pi n$)²⁰ Thus only photogenerated carriers near the surfaces (interfaces) are being monitored by TR and it provides much better surface sensitivity compared with conventional transient absorption spectroscopy (TA), which is highly sensitive to bulk carriers. Typically, it is the behavior of carriers near these interfaces or surfaces that are of the most interest for solar energy conversion. When carriers are present, they induce a change in the refractive index. At low

carrier density, the measured change in reflectance ($\Delta R/R$) and change in the real part of the refractive index n (Δn) has the following relation:²⁰

$$\frac{\Delta R}{R}(\hbar\omega) \cong \frac{4}{n(\hbar\omega)^2 - 1} \times \Delta n(\hbar\omega) \tag{1}$$

The change in refractive index $\Delta n(\hbar \omega)$ can be associated with a change of absorption coefficient $\Delta \alpha(\hbar \omega)$ by a Kramers-Kronig relationship,²²

$$\Delta n(\hbar\omega) = \frac{c}{\pi} P \int_{0}^{+\infty} \frac{\Delta \alpha(\hbar\omega)}{{\omega_1}^2 - {\omega}^2} d\omega_1$$
 (2)

Where c is the speed of light, and P is Cauchy principal value of the integral. At low carrier densities $\Delta \alpha$ can be linearly related to the density of charge carriers by²³

$$\Delta\alpha(\hbar\omega) = \alpha_0(\hbar\omega) \times \frac{N}{N_m} \tag{3}$$

Where $\alpha_0(\hbar\omega)$ is the absorption coefficient of the ground state, N is the carrier density induced by the light pulse and N_m is the saturation charge density and is on the order of $10^{19}cm^{-3}$. Equation 3 is valid under low excitation density N<<N_m. Thus, we find that the measured change in reflectance $\Delta R/R$ can be directly related to photogenerated charge carrier dynamics in the probing region near the surfaces. (See representative TR spectrum of example thin film MAPbI₃ solar cell and its Kramer-Kronig transformation in Figure 2a) For thin film samples and when the probe penetrates to the back surface an interference pattern can occur that contains information about both the surface carriers as well as bulk carriers. (Figure 2a) Typically the spectrum can be decomposed into a part that penetrates through the whole sample and a part that only contains wavelengths that have reflected from the surface, thus there is an opportunity to following both surface carrier dynamics simultaneously with bulk carrier dynamics. Figure 2b shows that from one experiment one can extract both bulk carrier dynamics (green-trace) and surface carrier dynamics (red-trace). Thus, from one measurement we can capture both bulk carrier dynamics and surface carrier dynamics.

At the surfaces of semiconductors, the dominant carrier dynamics are carrier diffusion away from the probing region and surface recombination. The diffusion of carriers away from the surfaces can be described by the one-dimensional diffusion equation:²⁴

$$\frac{\partial N(x,t)}{\partial t} = D \frac{\partial^2 N(x,t)}{\partial x^2} - \frac{N(x,t)}{\tau_B} \tag{4}$$

where N(x, t) is the carrier density as a function of depth (x) and time (t), D is the diffusion coefficient and τ_B is the bulk carrier lifetime. At t = 0, the carrier profile is described by Beer's law:

$$N(x,0) = N_0 \cdot e^{-\alpha_0 x} \tag{5}$$

where N_0 equals to the total number of absorbed photons and the recombination of carriers at surfaces can be described by:

$$\frac{\partial N(x,t)}{\partial t}|_{x=0} = \frac{S}{D}N(0,t) \tag{6}$$

where S is the surface recombination velocity (SRV). If the sample under investigation is a thin film material with thickness L, another equation is needed to describe surface recombination that can occur at the back or bottom surface.

$$\frac{\partial N(x,t)}{\partial t}|_{x=L} = -\frac{S_B}{D}N(L,t) \tag{7}$$

Combining equation 4-7, the surface carrier dynamics can be modelled, and diffusion constant D and S can be obtained. In an actual measurement, the sample is excited with laser pump pulses of several different wavelength. By doing so, the carrier generation profile N(x,0) changes as the pump penetration depth $\left(d_{pump} = \frac{1}{\alpha(\hbar\omega_{pump})}\right)$ varies with pump energy. The dynamics for change in reflectance $(\Delta R/R)$ near band-gap energy is then extracted for each pump wavelength and a global fitting procedure is performed based on equation 4-7 to extract D and S. (Figure 2c for representative $\Delta R/R$ dynamics and modelling with carrier diffusion, S)

Some recent studies have been done to measure surface carrier dynamics in solar cell materials with the use of TR. Yang et. al.²¹ measured the *D* and *S* for solution processed thin film MAPbI₃. The measured value for S is around 0.5 \times 10³cm s⁻¹, 3 orders smaller than un-passivated Si²⁵ or CdTe²⁶, making MAPbI₃ a good candidate for solar cell application. Also, by measuring both surface and bulk carrier dynamics, Yang et. al. were able to determine that the largest contribution to the carrier depopulation in thin MAPbI₃ films arises from the top and bottom surface recombination rather than from bulk or grain boundary defects.²¹ Wu et. al.²⁷ performed TR measurements on CsSnI₃ and the obtained S value of around 2 $\times 10^3 cm \, s^{-1}$ without any surface passivation. Based on value of S, they estimated surface trap density to be $\sim 10^{11}$ cm⁻² and proposed a relationship with grain size. Most recently, Tong et. al. 28 reported a decrease in S by a factor of 10 by adding GuaSCN to passivate MAPb_{0.4}Sn_{0.6}I₃ solar cells by the use of TR. These layers were then used in an all tandem perovskite solar cell achieving power conversion efficiencies greater than 20%. These studies demonstrate the power of TR in probing the ultrafast interfacial dynamics in solar cells and how these measurements can be used to guide materials development.

Surface carrier dynamics and transient electric field in solar fuel systems probed by transient photoreflectance spectroscopy

Apart from solar cells, photoelectrochemical (PEC) cells that are capable of directly converting solar energy to fuels are of great interest.²⁹⁻³¹ In PEC cells, the photoelectrodes generally consist of semiconducting materials such as Si, III-V semiconductors or metal oxides.³²⁻³⁴ At the electrode surfaces, a junction or an electric field can form as a result of fermi-level equilibration. The built-in electric field at the interface can help to separate photogenerated charges that drive solar-to-fuel conversion processes. Photogenerated electrons or holes transfer across the interface to participate in a reduction or oxidation reaction and holes or electrons are transported to the counter electrode via an internal or external circuit for the oxidation or reduction reaction. These types of charge transfer processes can occur within the ultrafast time scale.³⁵⁻³⁶ The strength and the dynamics of the interfacial electric field play a key role in governing the carrier dynamics at electrode surfaces. Static electrical and electrochemical techniques such as Mott-Schottky can

determine the strength of the built-in field. However, to investigate changes to the field on ultrafast timescales that occur due to carrier dynamics at the interface, a transient technique is needed such as transient photoreflectance (TPR) spectroscopy.³⁷

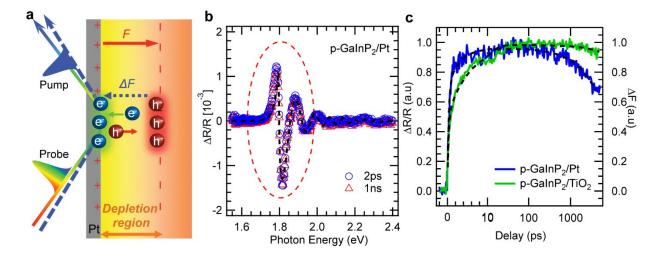


Figure 3. Surface transient electric field probed by transient photoreflectance **spectroscopy a.** Schematic illustration of transient photoreflectance spectroscopy at a ptype semiconductor/Pt junction. A broadband probe pulse spanning the semiconductor bandgap is reflected from an interface of interest. A pump pulse generates charge carriers that separate due to the presence of the depletion region. The movement of charge carriers changes the electric field at these interfaces. This change in electric field can be probed with broadband probe pulse and surface carrier dynamics can be followed by following the dynamics of the field recorded in the probing pulse. **b.** Transient spectra for p-GaInP₂/Pt junction at pump-probe delay of 2 ps (blue-circles) and 1 ns (red-triangles). An above bandgap monochromatic pump pulse modulates the reflectance, either via band filling due to the presence of free carriers or via surface field due to charge separation across the interface. Franz-Keldysh oscillations (FKO) is observed near the bandgap at 1.8 eV probe energy (red-dashed circle). The black-dash traces are the simulations of the FKO oscillation. The simulation matches the transient spectra. c. FKO kinetics at p-GaInP₂/Pt and p-GaInP₂/TiO₂ interfaces. The initial rise is an indication of carrier drift out of the depletion region producing a field in the opposite direction as the built-in field. The carrier lifetime is much longer in the GaInP₂/TiO₂ due to a barrier for both electrons and holes to recombine. Reproduced with permission from ref 37. Copyright 2015 American Association for the Advancement of Science

The working principle of TPR is similar to TR spectroscopy mentioned above. (See illustration in Figure 3a) It is also a pump probe-based technique that uses a white-light continuum as a probe that reflects off surfaces or interfaces of interest. The change in reflectance ($\Delta R/R$) of the broadband probe from a specific interface is monitored as a function of pump-probe delay. The difference between TR and TPR is in the transient

(9)

spectra ($\Delta R(\hbar\omega)/R$) and the presence or absence of a built-in electric field. For bare photoelectrode surfaces, only a small to no field is present prior to photoexcitation, but a transient surface field forms when electrons or holes are trapped at the surface and the other carrier diffuses away from the surface.³⁸ Since there is no (or very little) static surface field to begin with, only a small amount of charge carriers will separate and create a low surface field. In the low-field limit the response according to the electro-reflectance effect,³⁹ $\Delta R(\hbar\omega)/R$) should be proportional to the third derivative of the refractive index n ($\hbar\omega$) with respect to photon energy,

$$\frac{\Delta R}{R}(\hbar\omega) \sim \frac{\partial^3 n(\hbar\omega)}{\partial(\hbar\omega)^3} \tag{8}$$

For junctions formed at photoelectrode surfaces where there is a large intrinsic electric field (build-in field F) present, the situation changes. The static reflectance spectra (R) will exhibit periodic oscillations above the bandgap, known as Franz-Keldysh oscillations (FKO).⁴⁰ Under low-pump intensity, the photogenerated charge carriers separate due to the intrinsic field and causes a small perturbation (transient field ΔF) to the intrinsic field (F) which is represented in the measured transient spectra($\Delta R(\hbar\omega)/R$). According to Franz-Keldysh theory,⁴¹

$$\frac{\Delta R(\hbar\omega)}{R} \sim F^{\frac{1}{3}} \cdot \Delta F \cdot exp\left(\frac{-2 \cdot \Gamma \cdot \mu \cdot (E_g - \hbar\omega)}{e \cdot F \cdot \hbar}\right) \cdot A(F)$$

Where F is the intrinsic field, ΔF is the transient change in electric field, Γ is a phenomenological broadening parameter that damps the FKO amplitude for increasing energy, μ is the carrier reduced mass, E_g is the bandgap and A(F) contains Airy functions, exhibiting the oscillatory behavior. Such that for the case of junctions formed at photoelectrode surfaces, the transient spectra $(\Delta R/R)$ is proportional to the intrinsic field strength and change in field strength. (See Figure 3b for representative modelling of measured transient spectrum $\Delta R/R$ on example photoelectrode p-GaInP₂). The dynamics of the measured change in reflectance is proportional to the dynamics of the transient field ΔF , which can be used to determine the dynamics of surface charge separation,

recombination and charge transfer. (See Figure 3c for representative dynamics on the same junction)

Recent TPR studies probe surface carrier dynamics for typical photoelectrochemical systems. Yang et. al.³⁷ applied TPR to p-GaInP₂/Pt and p-GaInP₂/TiO₂ (H₂ evolution photoelectrode) junctions and determined that the surface carrier drift time is on the order of a few ps due to the build-in field. They found that upon adding an ALD amorphous TiO₂ layer a large field forms at the interface which is consistent with the formation of a p-n junction. The p-n junction helps to protect the p-GaInP₂ by keeping carriers away from the surface where photocorrision can occur. Electrons are accelerated into the TiO₂ layer where they are able to perform reduction reactions while holes are pushed away from the surface towards the back electrode where they can be swept out and eventually perform oxidation reactions. The field also helps to reduce the back-electron transfer prolonging the excited state lifetime and protects the surfaces by keeping the oxidizing holes away from the semiconductor/electrolyte surface. Cooper et. al. 42 studied BiVO₄ films (O₂ evolution photoelectrode) with TA and TPR and developed a model to describe charge carrier response at BiVO₄ surfaces. They discovered the principle spectra components were related to shifting and broadening of the ground-state optical properties for t < 10 ps. The TPR technique can be further applied to study interface carrier and field dynamics of many photoelectrode interfaces.

Ultrafast reaction intermediates in solar fuel systems probed by transient attenuated reflection spectroscopy

Another important aspect in the solar fuel process is to understand how chemical reactions occur on photoelectrode surfaces. The central task is to identify reaction intermediates of the fuel forming reaction cycle, especially the primary intermediates that occur immediately after the ultrafast charge transfer event. ⁴³⁻⁴⁶ The identification of such intermediates can provide insights into the identity of the active catalytic reaction sites and the potential energy landscape, which will enable the design of the next generation of stable, cheap and efficient catalyst. Infrared (IR) light is sensitive to molecular vibrations and transient IR has been widely used to identify reaction intermediates in a variety of

chemical reactions.⁴⁷⁻⁵⁰ For reactions at photoelectrode surfaces, surface sensitive ultrafast infrared probes such as sum frequency generation⁵¹ (SFG) or attenuated total reflection⁵² (ATR) have been developed. Both techniques are very important to the field of solar fuel research and SFG can in principle provide a higher surface sensitivity, down to a fraction of a monolayer, than does ATR. But due to the non-linear nature of SFG, it is difficult generate a signal strong enough to be detected. In the scope of this perspective, we will not detail ultrafast SFG techniques, instead, we will introduce *in-situ* transient attenuated total reflection spectroscopy in a photoelectrochemical cell where the photocatalytic reaction is actively occurring.

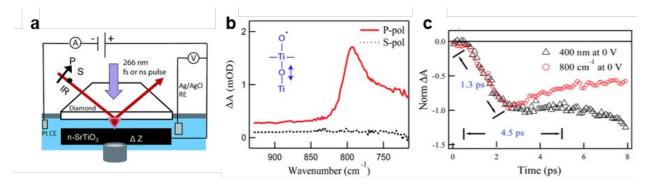


Figure 4. Ultrafast reaction intermediates probed by transient attenuated reflection spectroscopy a. Illustration of *in-situ* transient attenuated total reflection infrared spectroscopy for probing ultrafast reaction intermediates on example n-SrTiO₃ photoelectrode surface. The IR probe light incident on diamond prism with 45-degree angle. At the interface of prism and water, an evanescent wave is generated and probes the reaction intermediates on sample surfaces. The polarization of IR probe light can be controlled to have P and S polarization. P polarization has dipole perpendicular to the surfaces. CE, RE, and WE refer to the counter, reference, and working electrodes **b.** Representative oxyl radical intermediate observed by its vibrations perpendicular to the n-SrTiO₃ surface near 795 cm⁻¹ at 10 ps after photoexcitation. This ultrafast reaction intermediate has vibrational dipole perpendicular to the sample surface and can only be observed with p-polarized probe **c.** The ultrafast 1.3 ps formation dynamics for the oxyl radical described above. Reproduced with permission from ref 45. Copyright 2017 Royal Society of Chemistry

Transient attenuated total reflection infrared spectroscopy (TATR) is a pump-probe based technique that employs 100 fs pulses of infrared light as the probing light in an attenuated total reflection geometry. (See Figure 4a for example setup to probe reaction intermediates) A high energy pump pulse excites the sample to trigger the photo-driven reaction. Then after some time delay, the probe IR pulse with $\sim 150~\rm cm^{-1}$ width is focused into a prism made of an IR transparent material such as ZnSe, Diamond or MgF₂. The

polarization of the probe pulse is controlled by an IR polarizer with P polarization indicating an electric field perpendicular to the surfaces and S polarization indicating an electric field parallel to the surface. At the interface of the prism and reaction media, the IR light undergoes attenuated total reflection when the incident angle is bigger than the critical angle and an evanescent wave is generated at that interface with an exponential decaying intensity:⁵³

$$I = I_0 e^{-\frac{z}{d}} \tag{10}$$

where I_0 is the maximum intensity at the interface, z is the distance from the prism surfaces and d is the penetration depth defined by:⁵³

$$d = \frac{\lambda}{2\pi\sqrt{n_1^2 sin^2(\theta) - n_2^2}}$$
 (11)

where λ is the probe wavelength, n_1 is the real part of the refractive index of the prism, θ is the incident angle to surface normal at prism/electrolyte interface and n_2 is the real part of the refractive index for the electrolyte. In a typical measurement with 45-degree incident angle, d is on the order of 100 - 800 nm. In the TATR, the measured change is in absorbance (ΔA). For a weak absorbing medium, ΔA can be approximated as:⁵³

$$\Delta A = -\log\left(\frac{1 - T^{ON}}{1 - T^{OFF}}\right) \tag{12}$$

Where $T^{\rm on}$ is the transmitted IR light when pump is on and $T^{\rm off}$ is the transmitted light when pump is off. By analyzing the polarization dependent transient IR spectra ΔA in the desired region for specific molecular vibration, identities of ultrafast reaction intermediates can be determined. (see Figure 4b for one ultrafast reaction intermediate perpendicular to the surfaces) The kinetics associated with the spectra determines the formation dynamics of the reaction intermediates. (Figure 4c for the formation kinetics of reaction intermediate)

With the use of TATR spectroscopy, Herlihy et. al.⁵² discovered an oxyl radical on n-SrTiO₃ surfaces as an early intermediate for water splitting in the form of Ti-O·. It has a unique vibrational signature near 795 cm⁻¹ and it vibrates perpendicular to the surface (*i.e.*, it only

shows up when using a p-polarized IR-probe). Later, Chen *et. al.*⁵⁴ determined the formation dynamics of these types of reaction intermediates. They form in the ultrafast time scale with a 1.3 ps formation time. As the full water splitting reaction takes much longer to complete,⁵⁵ these ultrafast intermediates can stabilize photogenerated charges that improve the solar-to-fuel conversion efficiency. The TATR spectroscopic technique can be further developed and applied to many more solar fuel systems to discover the identity of early reaction intermediates.

Conclusion and Outlook

Interfacial carrier dynamics is a critical component in the overall solar-energy conversion process. A set of fundamental photoinduced processes including light-harvesting, charge separation, interface recombination, charge-carrier mobility, charge transfer and the ultrafast formation of surface reaction intermediates govern the process and efficiency of solar energy conversion. The use of ultrafast probes such as transient reflection (TR), transient photoreflectance (TPR) and transient attenuated total reflection (TATR) to investigate the interfacial dynamics in the solar energy conversion materials provides fundamental insights that can be used in the design and optimization of solar energy conversion systems. From these investigations, many crucial properties such as surface carrier recombination, surface carrier separation, surface field kinetics and reaction intermediates formation of solar energy conversion materials have been discovered. These emerging ultrafast spectroscopic techniques can be further applied to investigate specific properties of emerging solar energy conversion materials and systems. An intriguing avenue of research is to apply these techniques to *in-situ* characterization.

The field of ultrafast interfacial dynamics has progressed rapidly in recent years with many ultrafast probes being developed. One particularly important scientific challenge is to understand interfacial dynamics *in-situ*, for example, in operating solar cells or photoelectrochemical devices. The understanding of *in-situ* interfacial dynamics should provide links to the fundamental photoinduced processes and thus, overall device efficiency, as well as, photoelectrode stability. These methods allow for the elucidation of design principles that are currently lacking detail and specificity. In addition to

understanding *in-situ* dynamics, another crucial aspect that needs further investigation is the ultrafast structural dynamics that occur at interfaces during photochemical reactions. Many of the most promising solar energy conversion technologies involve complex materials, inorganic/organic hybrids and/or heterogeneous systems. Ultrafast interfacial structural fluctuation likely affects the long-term stability as well as the efficiency of solar energy conversion devices. Further efforts are therefore are needed to further develop probes that are capable to investigate ultrafast interfacial structural dynamics.

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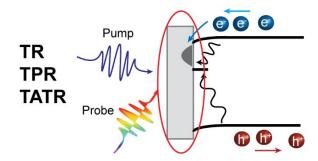
- 1. Lewis, N. S.; Nocera, D. G. Powering the planet: Chemical challenges in solar energy utilization. *Proc. Natl. Acad. Sci.* **2006**, *103* (43), 15729.
- 2. Nozik, A. J.; Miller, J. Introduction to Solar Photon Conversion. *Chem. Rev.* **2010,** *110* (11), 6443-6445.
- 3. Nocera, D. G. Solar Fuels and Solar Chemicals Industry. Acc. Chem. Res. 2017, 50 (3), 616-619.
- 4. Ponseca, C. S.; Chábera, P.; Uhlig, J.; Persson, P.; Sundström, V. Ultrafast Electron Dynamics in Solar Energy Conversion. *Chem. Rev.* **2017**, *117* (16), 10940-11024.
- 5. Blakers, A.; Zin, N.; McIntosh, K. R.; Fong, K. High Efficiency Silicon Solar Cells. *Energy Procedia* **2013,** *33*, 1-10.
- 6. Montoya, J. H.; Seitz, L. C.; Chakthranont, P.; Vojvodic, A.; Jaramillo, T. F.; Nørskov, J. K. Materials for solar fuels and chemicals. *Nat. Mater.* **2016**, *16*, 70.
- 7. Nocera, D. G. The Artificial Leaf. *Acc. Chem. Res.* **2012**, *45* (5), 767-776.
- 8. Lewis, N. S. Developing a scalable artificial photosynthesis technology through nanomaterials by design. *Nat. Nanotechnol.* **2016**, *11*, 1010.
- 9. Bushuyev, O. S.; De Luna, P.; Dinh, C. T.; Tao, L.; Saur, G.; van de Lagemaat, J.; Kelley, S. O.; Sargent, E. H. What Should We Make with CO2 and How Can We Make It? *Joule* **2018**, *2* (5), 825-832.
- 10. Gust, D.; Moore, T. A.; Moore, A. L. Solar Fuels via Artificial Photosynthesis. *Acc. Chem. Res.* **2009**, *42* (12), 1890-1898.
- 11. Johnston, M. B.; Herz, L. M. Hybrid Perovskites for Photovoltaics: Charge-Carrier Recombination, Diffusion, and Radiative Efficiencies. *Acc. Chem. Res.* **2016**, *49* (1), 146-154.

- 12. Shi, J.; Li, Y.; Li, D.; Luo, Y.; Wu, H.; Meng, Q. From Ultrafast to Ultraslow: Charge-Carrier Dynamics of Perovskite Solar Cells. *Joule* **2018**, *2* (5), 879-901.
- 13. Archer, M. D.; Nozik, A. J. *Nanostructured and photoelectrochemical systems for solar photon conversion*. Imperial College Press: London, 2010.
- 14. Schmidt, J.; Peibst, R.; Brendel, R. Surface passivation of crystalline silicon solar cells: Present and future. *Sol. Energy Mater. Sol. Cells.* **2018**, *187*, 39-54.
- 15. Chen, Q.; Zhou, H.; Song, T.-B.; Luo, S.; Hong, Z.; Duan, H.-S.; Dou, L.; Liu, Y.; Yang, Y. Controllable Self-Induced Passivation of Hybrid Lead Iodide Perovskites toward High Performance Solar Cells. *Nano Lett.* **2014**, *14* (7), 4158-4163.
- de Quilettes, D. W.; Vorpahl, S. M.; Stranks, S. D.; Nagaoka, H.; Eperon, G. E.; Ziffer, M. E.; Snaith, H. J.; Ginger, D. S. Impact of microstructure on local carrier lifetime in perovskite solar cells. *Science* **2015**, *348* (6235), 683.
- 17. Noel, N. K.; Abate, A.; Stranks, S. D.; Parrott, E. S.; Burlakov, V. M.; Goriely, A.; Snaith, H. J. Enhanced Photoluminescence and Solar Cell Performance via Lewis Base Passivation of Organic—Inorganic Lead Halide Perovskites. *ACS nano* **2014**, *8* (10), 9815-9821.
- 18. Stranks, S. D.; Eperon, G. E.; Grancini, G.; Menelaou, C.; Alcocer, M. J. P.; Leijtens, T.; Herz, L. M.; Petrozza, A.; Snaith, H. J. Electron-Hole Diffusion Lengths Exceeding 1 Micrometer in an Organometal Trihalide Perovskite Absorber. *Science* **2013**, *342* (6156), 341.
- 19. Beard, M. C.; Turner, G. M.; Schmuttenmaer, C. A. Transient photoconductivity in GaAs as measured by time-resolved terahertz spectroscopy. *Phys. Rev. B* **2000**, *62* (23), 15764-15777.
- 20. Yang, Y.; Yan, Y.; Yang, M.; Choi, S.; Zhu, K.; Luther, J. M.; Beard, M. C. Low surface recombination velocity in solution-grown CH3NH3PbBr3 perovskite single crystal. *Nat. Commun.* **2015**, *6*, 7961.
- 21. Yang, Y.; Yang, M.; Moore, David T.; Yan, Y.; Miller, Elisa M.; Zhu, K.; Beard, Matthew C. Top and bottom surfaces limit carrier lifetime in lead iodide perovskite films. *Nat. Energy* **2017**, *2*, 16207.
- 22. Chemla, D.; Miller, D.; Smith, P.; Gossard, A.; Wiegmann, W. Room temperature excitonic nonlinear absorption and refraction in GaAs/AlGaAs multiple quantum well structures. *IEEE J. Quantum Electron.* **1984**, *20* (3), 265-275.
- 23. Schmitt-Rink, S.; Chemla, D. S.; Miller, D. A. B. Theory of transient excitonic optical nonlinearities in semiconductor quantum-well structures. *Phys. Rev. B* **1985**, *32* (10), 6601-6609.
- 24. Hoffman, C. A.; Jarašiūnas, K.; Gerritsen, H. J.; Nurmikko, A. V. Measurement of surface recombination velocity in semiconductors by diffraction from picosecond transient free-carrier gratings. *Appl. Phys. Lett.* **1978**, *33* (6), 536-539.
- 25. Sabbah, A. J.; Riffe, D. M. Measurement of silicon surface recombination velocity using ultrafast pump–probe reflectivity in the near infrared. *J. Appl. Phys.* **2000**, *88* (11), 6954-6956.
- 26. Cohen, R.; Lyahovitskaya, V.; Poles, E.; Liu, A.; Rosenwaks, Y. Unusually low surface recombination and long bulk lifetime in n-CdTe single crystals. *Appl. Phys. Lett.* **1998,** *73* (10), 1400-1402.
- 27. Wu, B.; Zhou, Y.; Xing, G.; Xu, Q.; Garces, H. F.; Solanki, A.; Goh, T. W.; Padture, N. P.; Sum, T. C. Long Minority-Carrier Diffusion Length and Low Surface-Recombination Velocity in Inorganic Lead-Free CsSnI3 Perovskite Crystal for Solar Cells. *Adv. Funct. Mater.* **2017**, *27* (7), 1604818.
- 28. Tong, J.; Song, Z.; Kim, D. H.; Chen, X.; Chen, C.; Palmstrom, A. F.; Ndione, P. F.; Reese, M. O.; Dunfield, S. P.; Reid, O. G.; Liu, J.; Zhang, F.; Harvey, S. P.; Li, Z.; Christensen, S. T.; Teeter, G.; Zhao, D.; Al-Jassim, M. M.; van Hest, M. F. A. M.; Beard, M. C.; Shaheen, S. E.; Berry, J. J.; Yan, Y.; Zhu, K. Carrier lifetimes of >1 μs in Sn-Pb perovskites enable efficient all-perovskite tandem solar cells. *Science* **2019**, *364* (6439), 475.
- 29. Fujishima, A.; Honda, K. Electrochemical Photolysis of Water at a Semiconductor Electrode. *Nature* **1972**, *238*, 37.

- 30. Hu, S.; Shaner, M. R.; Beardslee, J. A.; Lichterman, M.; Brunschwig, B. S.; Lewis, N. S. Amorphous TiO2 coatings stabilize Si, GaAs, and GaP photoanodes for efficient water oxidation. *Science* **2014**, *344* (6187), 1005.
- 31. Khaselev, O.; Turner, J. A. A Monolithic Photovoltaic-Photoelectrochemical Device for Hydrogen Production via Water Splitting. *Science* **1998**, *280* (5362), 425.
- 32. Nielander, A. C.; Shaner, M. R.; Papadantonakis, K. M.; Francis, S. A.; Lewis, N. S. A taxonomy for solar fuels generators. *Energy Environ. Sci.* **2015**, *8* (1), 16-25.
- 33. Young, J. L.; Steiner, M. A.; Döscher, H.; France, R. M.; Turner, J. A.; Deutsch, Todd G. Direct solar-to-hydrogen conversion via inverted metamorphic multi-junction semiconductor architectures. *Nat. Energy* **2017**, *2*, 17028.
- 34. Kim, T. W.; Choi, K.-S. Nanoporous BiVO4 Photoanodes with Dual-Layer Oxygen Evolution Catalysts for Solar Water Splitting. *Science* **2014**, *343* (6174), 990.
- 35. Waegele, M. M.; Chen, X.; Herlihy, D. M.; Cuk, T. How surface potential determines the kinetics of the first hole transfer of photocatalytic water oxidation. *J. Am. Chem. Soc.* **2014**, *136* (30), 10632-9.
- 36. Wu, K.; Zhu, H.; Lian, T. Ultrafast Exciton Dynamics and Light-Driven H2 Evolution in Colloidal Semiconductor Nanorods and Pt-Tipped Nanorods. *Acc. Chem. Res.* **2015**, *48* (3), 851-859.
- 37. Yang, Y.; Gu, J.; Young, J. L.; Miller, E. M.; Turner, J. A.; Neale, N. R.; Beard, M. C. Semiconductor interfacial carrier dynamics via photoinduced electric fields. *Science* **2015**, *350* (6264), 1061.
- 38. Shen, H.; Dutta, M. Franz–Keldysh oscillations in modulation spectroscopy. *J. Appl. Phys.* **1995**, *78* (4), 2151-2176.
- 39. Aspnes, D. E. Direct Verification of the Third-Derivative Nature of Electroreflectance Spectra. *Phys. Rev. Lett.* **1972**, *28* (3), 168-171.
- 40. Pollak, F. H. Study of semiconductor surfaces and interfaces using electromodulation. *Surf. Interface Anal.* **2001,** *31* (10), 938-953.
- 41. Shen, H.; Pollak, F. H. Generalized Franz-Keldysh theory of electromodulation. *Phys. Rev. B* **1990**, 42 (11), 7097-7102.
- 42. Cooper, J. K.; Reyes-Lillo, S. E.; Hess, L. H.; Jiang, C.-M.; Neaton, J. B.; Sharp, I. D. Physical Origins of the Transient Absorption Spectra and Dynamics in Thin-Film Semiconductors: The Case of BiVO4. *J. Phys. Chem. C* **2018**, *122* (36), 20642-20652.
- 43. Walter, M. G.; Warren, E. L.; McKone, J. R.; Boettcher, S. W.; Mi, Q.; Santori, E. A.; Lewis, N. S. Solar Water Splitting Cells. *Chem. Rev.* **2010**, *110* (11), 6446-6473.
- 44. Nørskov, J. K.; Bligaard, T.; Hvolbæk, B.; Abild-Pedersen, F.; Chorkendorff, I.; Christensen, C. H. The nature of the active site in heterogeneous metal catalysis. *Chem. Soc. Rev.* **2008**, *37* (10), 2163-2171.
- 45. Chen, X.; Aschaffenburg, D.; Cuk, T. One-electron intermediates of water oxidation & the role of solvation in their stability. *J. Mater. Chem. A* **2017**, *5* (23), 11410-11417.
- 46. Godin, R.; Kafizas, A.; Durrant, J. R. Electron transfer dynamics in fuel producing photosystems. *Curr. Opin. Electrochem.* **2017**, *2* (1), 136-143.
- 47. Zheng, J.; Kwak, K.; Xie, J.; Fayer, M. D. Ultrafast Carbon-Carbon Single-Bond Rotational Isomerization in Room-Temperature Solution. *Science* **2006**, *313* (5795), 1951.
- 48. Snee, P. T.; Yang, H.; Kotz, K. T.; Payne, C. K.; Harris, C. B. Ultrafast Infrared Studies of the Reaction Mechanism of Silicon–Hydrogen Bond Activation by η5-CpV(CO)4. *J. Phys. Chem. A* **1999**, *103* (49), 10426-10432.
- 49. Zhang, M.; de Respinis, M.; Frei, H. Time-resolved observations of water oxidation intermediates on a cobalt oxide nanoparticle catalyst. *Nat. Chem.* **2014**, *6* (4), 362-7.
- 50. Roberts, S. T.; Ramasesha, K.; Petersen, P. B.; Mandal, A.; Tokmakoff, A. Proton Transfer in Concentrated Aqueous Hydroxide Visualized Using Ultrafast Infrared Spectroscopy. *J. Phys. Chem. A* **2011**, *115* (16), 3957-3972.

- 51. Nihonyanagi, S.; Yamaguchi, S.; Tahara, T. Ultrafast Dynamics at Water Interfaces Studied by Vibrational Sum Frequency Generation Spectroscopy. *Chem. Rev.* **2017**, *117* (16), 10665-10693.
- 52. Herlihy, D. M.; Waegele, M. M.; Chen, X.; Pemmaraju, C. D.; Prendergast, D.; Cuk, T. Detecting the oxyl radical of photocatalytic water oxidation at an n-SrTiO3/aqueous interface through its subsurface vibration. *Nat. Chem.* **2016**, *8* (6), 549-55.
- 53. Milosevic, M. *Internal reflection and ATR spectroscopy*. Wiley-Blackwell: Oxford, 2012.
- 54. Chen, X.; Choing, S. N.; Aschaffenburg, D. J.; Pemmaraju, C. D.; Prendergast, D.; Cuk, T. The Formation Time of Ti-O(*) and Ti-O(*)-Ti Radicals at the n-SrTiO₃/Aqueous Interface during Photocatalytic Water Oxidation. *J. Am. Chem. Soc.* **2017**, *139* (5), 1830-1841.
- 55. Takanabe, K. Photocatalytic Water Splitting: Quantitative Approaches toward Photocatalyst by Design. *ACS Catal.* **2017,** *7* (11), 8006-8022.

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Transient reflection, photoreflectance and attenuated total reflection spectroscopy are developed to understand the ultrafast interfacial dynamics of solar conversion materials.