

Cite this: *J. Mater. Chem. B*, 2025, 13, 3593

# A thermoresponsive PEG-based methacrylate triblock terpolymer as a bioink for 3D bioprinting†

Kaiwen Zhang,<sup>a</sup> Anna P. Constantinou,<sup>d</sup> Cathal O'Connell,<sup>ib bc</sup>  
Theoni K. Georgiou<sup>ib \*d</sup> and Amy Gelmi<sup>ib \*a</sup>

Thermoresponsive polymers have been extensively reported for their use in tissue engineering and drug delivery applications. They have a wide range of thermoresponsive and rheological properties controlled by their structural characteristics, such as composition and architecture. Here, the considerable potential of a PEG based, non-ionic triblock thermoresponsive copolymer, namely OEGMA<sub>300</sub><sub>13</sub>-*b*-BuMA<sub>22</sub>-*b*-DEGMA<sub>12</sub> as a bioink for 3D printing with cell encapsulation is identified. The rheological tests showed that the gel transition temperature is 8 °C with 35% w/w concentration in PBS. The printability and cytotoxicity of the thermoresponsive gel were characterised and compared with those of commercial thermoresponsive polymer Pluronic<sup>®</sup>F127 in detail. Specifically, the 35% w/w triblock copolymer presented great printability with a printing speed of 450 mm min<sup>-1</sup> at 37 °C, and was less cytotoxic than F127 at both 20% and 30% w/w concentrations. A one-layer structure of human mesenchymal stem cell (hMSC) embedded triblock copolymer was successfully printed onto a glass slide at 37 °C. This provides an option to create a scaffold for stem cell culture and programming for further tissue engineering applications *via* direct printing of a cell-laden thermoresponsive polymer.

Received 16th November 2024,  
Accepted 10th February 2025

DOI: 10.1039/d4tb02572e

rsc.li/materials-b

## 1. Introduction

A thermoresponsive polymer is a smart material that responds to temperature fluctuation by changing its compatibility with a solvent and this class of smart materials has been widely studied because of their wide range of bio-applications, such as drug delivery, gene delivery, and tissue engineering.<sup>1–3</sup> Generally, thermoresponsive polymers in aqueous media can turn from hydrophilic to hydrophobic, or from hydrophobic to hydrophilic upon heating to specific temperatures; these are known as lower critical solution temperature (LCST) or upper critical solution temperature (UCST) polymers, respectively. If the polymer design is chosen appropriately, and under suitable external conditions, a thermoresponsive polymer solution will form a gel upon temperature changes as the liquid polymer solution undergoes gelation. For most clinical applications, LCST-type gels are preferred, as they are optimised to be in the solution state at lower temperature before injection, and then form a gel at body temperature (37 °C).

This behaviour has driven the development of polymer systems that can be used to effectively harness thermoresponsive properties. In the field of drug delivery, a polymer solution can be easily mixed with a drug at a lower temperature outside the body. Once injected into the body, it transforms into a gel, encapsulating the drug.<sup>4</sup> Depending on the polymer structure, the gel would degrade or dissolve in the body to release the drug in a controlled manner, achieving local release to minimise the side effect of systemic release.<sup>5</sup> For *in vivo* tissue engineering application, thermoreversible hydrogels work similarly to drug delivery agents by encapsulating cells rather than drug compounds. After injection, with body temperature higher than LCST, a gel forms and traps the cells, supporting cell proliferation and tissue formation while the gel degrades. With this, a new tissue could be generated to heal the defect site or replace the dysfunctional one.<sup>5</sup> Thermoresponsive gels (TRGs) can also be used as scaffold materials, supporting *in vitro* tissue engineering applications such as 3-dimension (3D) cell culturing or forming new tissue *in vitro* for transplantation. Particularly, TRGs are commonly used for sacrificial printing, such as hyaluronic acid – poly(*N*-isopropylacrylamide) (HA-pNIPAAm),<sup>6,7</sup> or Pluronic<sup>®</sup> F127 (F127), a commercialised thermoresponsive triblock copolymer. Upon heating, F127 changes from the liquid to gel state at physiological temperatures, which also highly depends on its concentration and solution ionic strength. As many studies reported potential inflammation or tissue irritation with F127 under certain conditions, cells are typically not encapsulated in the F127 itself for printing due to its potential cytotoxicity.<sup>8</sup> For

<sup>a</sup> School of Science, RMIT University, VIC, 3000, Australia.

E-mail: amy.gelmi@rmit.edu.au

<sup>b</sup> School of Engineering, RMIT University, Melbourne, VIC, 3000, Australia<sup>c</sup> Aikenhead Centre for Medical Discovery, St Vincent's Hospital Melbourne, Fitzroy, VIC, 3065, Australia<sup>d</sup> Department of Materials, Imperial College London, London, UK.

E-mail: t.georgiou@imperial.ac.uk

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4tb02572e>

sacrificial printing, F127 can be printed with another more biocompatible material for encapsulating cells, such as gelatin methacrylate (GelMA) or alginate.<sup>9,10</sup>

Directly printing a cell-encapsulated scaffold with a TRG can help simplify the process significantly and collect final cell products easily *via* temperature control, which requires hydrogels demonstrating great printability and cytocompatibility. However, this remains a challenge due to a lack of appropriate materials.<sup>11</sup> Only two TRGs that have reached clinical trials, Pluronic<sup>®</sup> F127 and Regel<sup>®</sup>, are not suitable, as the former presents cytotoxicity and the latter one requires UV exposure to stabilise the pattern, which could induce cell death.<sup>12</sup> The potential of a biodegradable thermoresponsive waterborne polyurethane (PU) gel as a bioink has been investigated, reporting the stacking layer and structure maintenance was limited, resulting in difficulties for complex structure printing.<sup>13,14</sup> Encapsulating cells within a bioink to be printed onto 3D structures allows 3D cell culture or forming new tissue with intercellular communication and bonding. Furthermore, the final products can be easily collected by removing the bioink *via* decreasing the temperature to lower than the gel point.

Recently, human mesenchymal stem cells (hMSCs) have been the commonly used cell type in 3D printing for tissue engineering applications, due to their extraordinary self-renewal and multipotent differentiation properties.<sup>15</sup> At first, a 3D printed scaffold for stem cell culture could highly increase the efficiency of autologous cell expansion with no cell loss,<sup>16</sup> which is critical when a large number of cells are required for the tissue engineering application, whereas a low number of cells are collected from patients. Secondly, it can support hMSCs to form networks and structures in 3D with appropriate extracellular matrix properties such as stiffness, mimicking *in vivo* conditions, which significantly affects cell differentiation and holds great promise for artificial organ creation and regenerative medicine applications.<sup>17</sup>

Here, we introduce a polyethylene glycol (PEG)-based triblock thermoresponsive polymer OEGMA300<sub>13</sub>-*b*-BuMA<sub>22</sub>-*b*-DEGMA<sub>12</sub> (OBD)<sup>18,19</sup> with promising thermoresponsive properties and improved non-cytotoxicity compared to F127 to be potentially used as a bioink for 3D printing. OEGMA300, BuMA, and DEGMA represent oligo(ethylene glycol) methyl ether methacrylate with 300 g mol<sup>-1</sup> molar mass, *n*-butyl methacrylate, and di(ethylene glycol) methyl ether methacrylate, respectively. The printability and cytotoxicity of this triblock terpolymer were characterised using hMSCs in this study, investigating the ability of this thermoresponsive polymer as a printable ink and as a bioink carrying human stem cells. This polymer is advantageous compared to the commercially available counterparts that it is easier to be functionalised and/or tailored because it is fabricated using a living polymerisation method that enables the precise control of the comonomer position and ratio.

## 2. Methods and materials

### 2.1 Materials

Pluronic<sup>®</sup> F127 and phosphate-buffered saline (PBS) were purchased from Sigma-Aldrich, Australia.

Triblock thermoresponsive polymer OEGMA300<sub>13</sub>-*b*-BuMA<sub>22</sub>-*b*-DEGMA<sub>12</sub> was in-house synthesised *via* sequential group transfer polymerisation (GTP), as previously described.<sup>18,19</sup> Briefly, 10 mg of the GTP catalyst, tetrabutylammonium bibenzoate, was added in a round bottom flask, followed by sealing with a rubber septum and purging with argon. 59 mL of freshly purified solvent, tetrahydrofuran (THF) and 0.6 mL of the GTP initiator, methyl trimethylsilyl dimethylketene acetal, (0.5 g, 3 mmol), were syringed into the flask. Each block was synthesised *via* sequential addition of the corresponding monomer, which was left to react for 10–15 minutes. The monomers used are: OEGMA300 (50 vol% solution in THF, 17 mL, 8.9 g, 30 mmol), BuMA (8.7 mL, 7.8 g, 55 mmol), and DEGMA (5.4 mL, 5.5 g, 29 mmol). After the exothermic reaction ceased, two aliquots of 0.1 mL were withdrawn for gel permeation chromatography (GPC) and proton nuclear magnetic resonance (<sup>1</sup>H NMR) spectroscopy analysis. OBD was purified *via* precipitation in cold *n*-hexane, followed by vacuum-drying for 10 days at room temperature. The experimental number-average molar mass, molar mass distribution, and composition were determined to be 9400 g mol<sup>-1</sup>, 1.17 and OEGMA300<sub>13</sub>-*b*-BuMA<sub>22</sub>-*b*-DEGMA<sub>12</sub> (OBD) at 42–33–25% w/w, respectively. The GPC and <sup>1</sup>H NMR spectroscopy results of OBD and its linear precursors are shown in Fig. S1 (ESI<sup>†</sup>). These confirm the successful sequential polymerisation, leading to an increase in molar mass values, and appearance of additional peaks corresponding to the newly added repeated units.

### 2.2 Polymer solution preparation

Pluronic<sup>®</sup> F127 solutions were prepared in PBS at 20% and 30% w/w concentrations. OBD solutions were prepared in PBS at 10, 15, 25, and 35% w/w. Solutions were agitated using a magnetic spinner at 300 rpm at room temperature for 24 h until fully dissolved. All polymer solutions were stored at 4 °C until use.

### 2.3 Thermoresponsive properties – visual tests

The thermoresponsive properties of 35% w/w OBD solutions in PBS were determined by observing the phase changes of the solutions from 23 to 80 °C, at a step of 1 °C, using a basic stirrer hotplate (IKA RCT) and a continuously stirred water bath. The temperature was measured using a thermometer, and the observation was recorded 2 min after the controller reached the set temperature to ensure that the solution has reached the desired temperature.

### 2.4 Rheological characterisation

The rheological properties of the 35% w/w OBD were tested with temperature sweep, shear rate ramp and shear recovery. A Physica MCR502 rheometer (Anton Paar, Graz, Austria) with 15 mm conical plate geometry (1° angle) was used for the measurement. Rheometer calibration at room temperature was conducted before testing samples to improve the accuracy of the results. Three independent measurements were repeated for each rheological test.



Temperature sweep measurements were performed at 0.1% strain and 1 rad s<sup>-1</sup> frequency, at a temperature range from 4 to 37 °C with a temperature increase rate of 2 °C min<sup>-1</sup>. Changes of shear storage modulus ( $G'$ ) and shear loss modulus ( $G''$ ) were measured during this temperature sweep. The gel point was defined as the temperature at which  $G'$  overcomes  $G''$  at the first time.<sup>20–22</sup> With the shear rate ramp test, viscosity of samples was tested at shear rate ramp from 0.01 to 1000 s<sup>-1</sup> at 37 °C (printing temperature). The yield stress was determined at which a viscosity peak is observed. In the shear recovery test, repetitive low strain of 0.1% for 40 s, followed by a high strain of 100% for 40 s was applied to samples at 37 °C and the change of  $G'$  over time was monitored.

## 2.5 Printability

The printability test followed the bio-ink characterisation protocol published by Connell *et al.*<sup>23</sup> using an INKREDIBLE+ 3D printer (CELLINK, Sweden). Samples were added to a 3 mL syringe, assembled with a 250 μm nozzle. Then the syringe was loaded to the pneumatic cartridge of the 3D printer and connected with an air pressure regulator. After heating the printing head to 37 °C and wrapping it with tin foil to stabilise the temperature of the sample, the bio-ink was ready to be extruded/printed. The printing G-code is listed in S5–S7 (ESI†).

**2.5.1 Extrusion rate test.** The extrusion rate test, or the line printing test, is carried out to determine the ideal writing speed by printing a few lines with selected pressure and specific concentration of OBD in different writing speeds. Matching the nozzle writing speed with the polymer extrusion rate is critical for steady and continuous printing, while increasing the moving speed of the nozzle can improve the efficiency of printing. The moving speeds were set to 300, 450, 600, 750, 900, and 1050 mm min<sup>-1</sup>. Once the bioink extruded steadily, a pattern was printed onto the glass slide. The pattern was imaged with a microscope. Fiji ImageJ was used to measure the width of the printed pattern.<sup>24</sup> Extrusion rate  $Q$  was calculated with eqn (1):

$$Q = \frac{\pi r^2 L}{2t} \quad (1)$$

The printed pattern is treated as a half cylinder. Where  $r$  is the radius of the cylinder and equal to half of the width of the printed line.  $L$  is the length of the printed line within time  $t$ , which is equal to the writing speed of the nozzle head.

**2.5.2 Layer stacking test.** Using optimised printing pressure and writing speed from the extrusion rate test, the layer height for 2-layer printing was set to 0.2 mm, while for multi-layer printing, the layer height was optimised accordingly. The printed pattern was kept warm (37 °C) to ensure the gel state of the polymer and help maintain the pattern structure. Once the printing was complete, an EVOS M5000 Imaging System was used to capture an image showing at least four pores. Printability ( $Pr$ ) with the stacking test was determined using eqn (2):

$$Pr = \frac{L^2}{16A} \quad (2)$$

where  $L$  is the internal perimeter of the pore and  $A$  is the

internal area of the pore.  $Pr = 1$  represents a defined square shape, showing great printability.  $Pr > 1$  indicates an irregular shape mainly due to over-gelation.  $Pr < 1$  indicates a liquefied, under-gelated bio-ink.

## 2.6 Stem cell culture and viability

**2.6.1 Stem cell culture.** Primary bone marrow derived human mesenchymal stem cells (hMSCs, Lonza, Switzerland) were cultured under standard cell culture conditions (37 °C, humidified atmosphere, 5% CO<sub>2</sub>). hMSCs were expanded in hMSC growth medium (Lonza, Switzerland) and grown to 80–90% confluency in standard cell culture flasks (Corning, USA).

**2.6.2 Leaching viability.** The cytotoxicity of the polymer gels was initially assessed using 15% and 35% w/w OBD in PBS. 60 μL of the polymer solution was gelled as a discrete droplet in a tissue culture well plate (Corning, USA). hMSCs were seeded at a density of 3000 cells per cm<sup>2</sup> into the well plate, in hMSC growth medium (Lonza, Switzerland). 24 h after cell seeding, the hMSCs were incubated with NucBlue™ Live ReadyProbes™ (Hoechst 33342) (Invitrogen™, USA) and NucGreen™ Dead 488 ReadyProbes™ (SYTOX™ Green) (Invitrogen™, USA) for 30 min at 37 °C. Cells were imaged using an EVOS M5000 Imaging System, and the ratio of live to dead hMSCs was calculated. Cells seeded on tissue culture plate (TCP) without OBD droplets served as the control group.

**2.6.3 Direct contact viability.** The viability of hMSCs grown directly on top of the 15% and 35% w/w OBD was assessed, with F127 20% and 30% w/w as comparison. 10 μL of the polymer was pipetted into each well of a 96 well plate (Corning, USA), and stored at 4 °C for 10 min to evenly coat the bottom of the wells with the polymer prior to gelling. Prior to cell seeding, the plate was incubated at 37 °C to form the gel. hMSCs were incubated with 10 μM CellTracker™ Orange CMRA Dye (Invitrogen™, USA) for 30 min before seeding onto the polymer at a density of 3000 cells per cm<sup>2</sup>. At 24, 48, and 72 h post seeding, hMSCs were counter-stained with NucGreen to indicate dead cells. After incubation for 30 min at 37 °C, cells were imaged using an EVOS M5000 Imaging System, and the ratio of live to dead hMSCs was calculated.

**2.6.4 Reseeding viability.** The viability of hMSCs post polymer seeding was also assessed. hMSCs were seeded on polymers, using the method described above, without any staining. 48 h after seeding, 50 μL cool media was added for 5 min to liquify and dilute the polymer. After removal of the media, TrypLE (Gibco™, America) was added to dissociate cells in the well plate. Then the cells were transferred to a new 6 well plate (Corning, America) with media topped up to 2 mL. The viability test was conducted 48 h after reseeding using NucBlue and NucGreen.

**2.6.5 Gel capsulation viability.** The hMSCs pellet containing 25 000 cells stained with CellTracker was collected after centrifugation. 20 μL of the cool polymer was mixed with the cell pellet. Two droplets of 10 μL solution were deposited into a well plate, and incubated at 37 °C for 5 min to solidify the polymer/cell droplets. The droplets were then immersed in complete cell media and incubated for 4 days, after which NucGreen stain was added to stain any dead cells.



## 2.7 Printing with cells

**2.7.1 Droplet printing with cells.** A pipette was used to print a few droplets of the cell-laden polymer onto a 12 well plate to mimic the 3D printing process, shown in Fig. 1. The cells were initially incubated with CellTracker Orange CMRA, as per instructions, to fluorescently label the hMSCs. A pellet of 25 000 cells was then collected after centrifugation and mixed with 20  $\mu\text{L}$  of the cool polymer. 10  $\mu\text{L}$  of the cell/polymer solution was pipetted into the 12 well plate, then incubated at 37  $^{\circ}\text{C}$  for 5 min to solidify the polymer/cell droplets. The droplets were then immersed in complete cell media. Before imaging with a confocal microscope, NucGreen stain was added to stain the dead cells. Confocal imaging ( $20\times$ ) with Z stack allows a 3D view of how the cells are distributed inside the polymer after printing.

**2.7.2 3D bioprinting.** A cell pellet with 250 000 hMSCs (pre-stained with CellTracker Orange CMRA) was mixed with 1 mL cool OBD, which was still under liquid conditions. The cell-laden bioink was then transferred to the syringe and incubated at 37  $^{\circ}\text{C}$ . For printing, the syringe was assembled to the 3D printer with the method described in Section 2.5. A one-layer pattern was printed into a 6 well plate using the optimised constant writing speed of 450  $\text{mm min}^{-1}$ . 2 mL of warm basal medium was then added to the well plate and then returned to the incubator at 37  $^{\circ}\text{C}$ . The EVOS M5000 Imaging System was used to image the cells inside the bioink 1 h post printing.

Combining the materials' rheological properties with printing parameters, the estimated shear stress that cells experienced during printing could be calculated with eqn (3) and (4):

$$\text{Shear strain } \gamma = \frac{4Q}{\pi R^3} \quad (3)$$

$$\text{Shear stress } \tau = \frac{4\eta Q}{\pi R^3} \quad (4)$$

where  $R$  is the radius of the printing nozzle, 125  $\mu\text{m}$ . Extrusion rate  $Q$  was calculated from eqn (1). Viscosity  $\eta$  of the sample was obtained from the shear rate test discussed in rheology characterisation in Section 2.4, based on the shear strain  $\gamma$  calculated using eqn (3).

## 2.8 Statistical analysis

Data presented here are expressed as column plots with mean  $\pm$  standard deviation (SD). A Welch's  $t$  test for cell viability was conducted by GraphPad Prism 9.

## 3. Results and discussion

### 3.1 Material properties

**3.1.1 Thermoresponsive properties.** The optimal printing parameters were determined from the preliminary printability experiment (Table S1, ESI<sup>†</sup>), where 35% w/w OBD at 75 kPa printing pressure demonstrated suitable filament production, extruding the steady filament at a reasonable speed. Hence these parameters were selected for all further tests.

The thermal properties of 35% w/w OBD were first determined by the visual test. At this concentration, polymer solutions were in the liquid state at 4  $^{\circ}\text{C}$ . At room temperature (23  $^{\circ}\text{C}$ ), polymer solution already transitioned to a clear gel condition. When the temperature increased to 39  $^{\circ}\text{C}$ , the gel became cloudy, indicating increased hydrophobic interactions. Gel syneresis, the process where the solvent is expelled from the polymer network due to internal stresses, began at 45  $^{\circ}\text{C}$  and progressed to a precipitation state by 50  $^{\circ}\text{C}$ . This indicates that viscosity of polymer solution increased as the temperature increased from 4  $^{\circ}\text{C}$  to room temperature, forming a gel. The exact gel point of 35% w/w OBD was not determined by the visual test as it already reached the gel state at room temperature and was thus determined by rheology temperature sweep tests.

**3.1.2 Rheology.** The rheological properties of a 35% w/w OBD solution were characterised using the temperature sweep test, shear rate ramp test, and shear recovery test. In Fig. 2, the shear storage modulus ( $G'$ ) is represented by red circles, while the shear loss modulus ( $G''$ ) is indicated by blue squares. During the temperature sweep test, both  $G'$  and  $G''$  began to increase after 6  $^{\circ}\text{C}$  (Fig. 2A). The temperature at which  $G'$  surpassed  $G''$  was identified as the gel point, occurring around 8  $^{\circ}\text{C}$ , which is significantly below room temperature. Maintaining the sample at a cool temperature helps preserve its liquid-like state, making it easier to handle before use. The shear rate ramp test conducted at 37  $^{\circ}\text{C}$  (the temperature used for cell printing) demonstrated the shear-thinning behaviour of the 35% w/w OBD, with a peak observed at a shear rate of 0.1  $\text{s}^{-1}$ , defined as the shear rate achieving yield stress (Fig. 2B). Beyond this point, the viscosity decreased as the shear rate increased, and the sample began to deform plastically. In the shear recovery test at 37  $^{\circ}\text{C}$  (Fig. 2C), following a high strain of 100% for 40 s,  $G'$  dropped significantly, indicating that the gel was thoroughly disrupted. The disruption of the thermogel by application of high shear strains is expected, as the polymer network is held by physical interactions. When the strain was reduced to 0.1%,  $G'$  returned to its initial values in around 20 s, an indication of the reversibility of our material, an

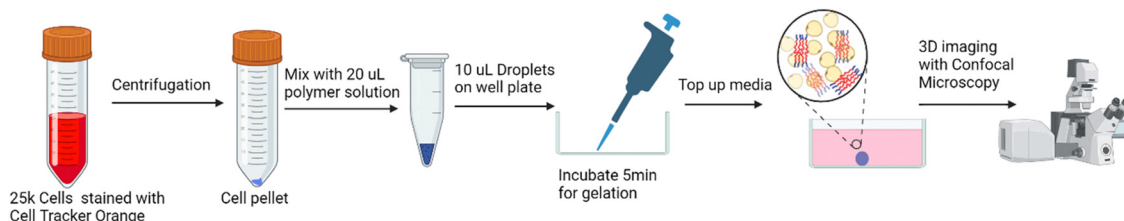


Fig. 1 Schematic diagram of the droplet printing test process.





Fig. 2 Rheology result of 35% w/w OBD (A) temperature sweep, (B) shear rate ramp, and (C) shear recovery. Red circles and blue squares represent shear storage modulus (elastic modulus,  $G'$ ) and shear loss modulus (viscous modulus,  $G''$ ), respectively. Storage modulus sharply decreases at high strain and recovers quickly (around 20 s) after removing strain.

advantage of physically crosslinked networks. After three repetitions of the high strain, the sample retained 85% of its original shear properties, showing minimal hysteresis.

### 3.2 Printability

**3.2.1 Line test – suitable printing speed.** The printability of commercialised product F127 itself or a mixture with other compounds such as GelMA, was widely reported in ranges of journals.<sup>6,10,25</sup> Our preliminary line printing experiment of F127 with 25% w/w concentration also proved the good quality printing (S3, ESI†). Therefore, the printability test here mainly focused on the 35% w/w OBD.

During the line test, the 35% w/w OBD was printed out on the glass slide with speed increased from  $300 \text{ mm min}^{-1}$  to  $1050 \text{ mm min}^{-1}$  (Fig. 3A, Table 1). When the printing speed was slower than  $600 \text{ mm min}^{-1}$ , OBD performed a steady extrusion. The average width of the printed pattern at  $300 \text{ mm min}^{-1}$  ( $435 \mu\text{m}$ ) was higher than that at  $450 \text{ mm min}^{-1}$  ( $401 \mu\text{m}$ ), showing that slower printing speed leads to a thicker pattern at the same extrusion rate. The average width of the pattern was slightly larger than nozzle size ( $250 \mu\text{m}$ ), which is attributed to the polymer liquefaction after exposing to a high shear rate during printing. The ideal writing speed was  $450 \text{ mm min}^{-1}$ , being used for further printing experiments, as it was the fastest printing speed with steady extrusion. Extrusion rate Q

at a  $450 \text{ mm min}^{-1}$  printing speed of 35% w/w OBD under 75 kPa was  $0.471 \mu\text{L s}^{-1}$ , calculated using eqn (1).

**3.2.2 Layer stacking test.** The printed lines of 35% w/w OBD stacked on top of one another (2-layer, shown in S4, ESI†), indicating a pass on layer stacking test. The average printability calculated with eqn (2) is 0.874, slightly less than 1. This could be attributed to the material liquification during high shear rate printing.

The printing process and pattern of 4-layer stack printing are shown in Fig. 3B and C. The stacking of multiple layers can be observed clearly. It is worth mentioning that with additional layer printing, the bioink might have difficulty adhering to the bottom pattern at the turnaround point. This issue can be improved by adjusting the Z height or reducing the writing speed. Since the 35% w/w OBD was successfully printed with multiple layers, it opens up the possibility of printing more complex structures for cell culture scaffolds. However, optimising the Z height for each layer is necessary, and the gel structure is required to fully recover after printing each layer before proceeding to the next layer.

### 3.3 Cell viability

**3.3.1 Leaching viability test.** The cell viability test focused on 35% w/w OBD due to its great printability, while OBD with 15% w/w concentration was also tested as it demonstrated



Fig. 3 Printability test of 35% w/w OBD. (A) Microscopy image of the line test pattern. From left to right (labelled as 1 to 6), the writing speed increased from  $300 \text{ mm min}^{-1}$  to  $1050 \text{ mm min}^{-1}$ . Scale bar, 2 mm. (B) Images of the 4-layer stacking test, the printing process from the 1st layer (i) to the 4th layer (iv). Scale bar 5 mm. (C) 4-layer pattern characterised by microscopy. Scale bar 2 mm.



Table 1 Line test results with writing speed from 300 to 1050 mm min<sup>-1</sup>

Line	1	2	3	4	5	6
Speed (mm min <sup>-1</sup> )	300	450	600	750	900	1050
Width (μm)	435 ± 16	401 ± 8	343 ± 30	328 ± 60	328 ± 30	267 ± 63

promising thermoresponsive properties, maintaining a stable gel state in the temperature range from 31 °C to 39 °C.<sup>18,19</sup> Concentrations of 20% and 30% w/w of the commercial product F127 were selected for comparison, ensuring its reversible thermal transition (minimal 15% w/w),<sup>26</sup> while maintaining a manageable viscosity.

Cell viability for hMSCs seeded around droplets of OBD (15% or 35% w/w) is shown in Fig. 4A. The relative viability of cells in both concentration setting was above 99%, indicating that the OBD material did not leach any cytotoxic substances.

**3.3.2 Direct contact viability test.** The normalised cell viability seeded directly on the polymer is shown in Fig. 4B. After 24 h of cell seeding on the polymer gel, at least 98% cell viability was maintained. 48 h post seeding, cell viability reduced slightly, still maintaining above 87% in most groups, while F127-20% showed a relatively low viability of 85%. Cell viability reduced further with a longer culture time with F127 at both concentrations presenting a relatively high cytotoxic effect, reducing to 35% at 20% w/w, which shows a significant difference compared to the control group. Meanwhile, cell viability in the OBD sample maintained at around 80%, showing no significant difference with the control group. There was also no significant difference between two different concentration settings in both types of polymers. In previous work, hepatic carcinoma HepG2 (human liver) cell viability in 20% w/w F127 was only 54% after 1 day of seeding, reducing further to 12% after 3 days of seeding;<sup>8</sup> 80% of MG-63 osteoblast-like cells cultured in 20% w/w F127 survived after 2 days with viability reducing to 60% in 4 days in culture.<sup>27</sup> As previously shown, ARPE-19 cells treated with either OBD or

F127 solutions across a range of concentrations (50–250 μg mL<sup>-1</sup>) presented significantly higher viability with OBD compared to F127.<sup>18</sup>

After culturing on the different polymers for 2 days, hMSCs were collected and reseeded, and their viability was further assessed (Fig. 4C). There was no significant difference between groups, with cell viability remaining above 84%. Culturing cells with OBD did not affect cell viability in further application, demonstrating that the OBD could be used as a 3D cell culture scaffold to allow high density cell culture.

In summary, OBD was non-cytotoxic to the hMSCs and demonstrated a higher cell viability than the commercialised product F127.

### 3.4 OBD printing with cells

**3.4.1 Droplet printing.** Droplet printing serves as an initial experiment to test 3D cell-laden polymer printing on a smaller scale. A few droplets of OBD mixed with cells were extruded using a pipette tip into a well plate to simulate the printing process with a minimal amount of the sample. Given its excellent printability, 35% w/w OBD was chosen for the droplet printing before proceeding with large-scale sample preparation for 3D cell printing.

Two hours after printing, the cells were distributed throughout the polymer droplet at different heights, as evidenced by the nuclei (blue) features in the 3D image (Fig. 5A). After 24 hours, most cells were clustered on the same plane (Fig. 5B). This distribution likely occurred because the polymer gel lacked binding sites for the cells, leading them to attach to the well plate substrate. Incorporating a block of arginine-glycine-aspartate



Fig. 4 Cell viability test of 15% and 35% w/w OBD, 20% and 30% w/w F127. (A) Leaching viability of cells seeded around 15% and 35% w/w OBD droplets after incubation for 48 h, with cell seeding on the tissue culture plate (TCP) only as the control group. (B) Direct contact viability of cells on the polymer with CellTracker. (C) Reseeding viability of cells seeded on the polymer for 2 days then reseeded on the well plate. Each data point represents the average of triplicate samples and normalised data (to TCP control) were presented as mean ± SD. \* indicates that there is statistical significance as measured by Welch's *t* test ( $p \leq 0.05$ ).





Fig. 5 Confocal images of cell embedded in 35% w/w OBD 2 h (A) and 24 h (B) after droplet printing. Nuclei was stained with NucBlue showing blue colour and the whole cell was stained with Cell tracker showing orange colour. Scale bar, 2 mm. (C) Cell viability of droplet printing. Each data point represents the average of triplicate samples, and the data are presented as mean  $\pm$  SD. Ns indicates there is no statistical significance as measured by Welch's t test ( $p > 0.05$ ).

(RGD peptide) could improve cell adhesion, though it might impact the physical properties of the polymer.<sup>28</sup>

On Day 4, the polymer could no longer maintain its droplet shape, likely due to the gel dissolving into the media, which resulted in cell spreading on the substrate. The viability of cells embedded in 35% w/w OBD with droplet printing remained above 90%. In summary, 35% w/w OBD successfully passed the droplet printing test while maintaining high cell viability.

**3.4.2 3D bioprinting.** 35% w/w OBD was mixed with hMSCs and printed at 75 kPa with 450 mm min<sup>-1</sup> writing speed *via* a 3D printer. From Fig. 6, the CellTracker stained cells can be clearly observed within the pattern region, showing successful cell printing. In this process, cells surrounded by high concentration OBD experienced osmotic pressure in the syringe for a long time (mins to hrs) before being extruded, followed by shear stress during extrusion. Both of these factors may impact cell viability. The interface between printed structures with the glass substrate is inexact, which might be due to liquefaction of the gel induced by the shear stress. With cells embedded, the rheological and gelation properties of the ink might be affected, resulting in the indistinct printed shape.<sup>7,29</sup>

The shear stress  $\tau$  that cells embedded in 35% w/w OBD experienced during printing was around 79 Pa, calculated with

eqn (1)–(4). A shear stress of 79 Pa is considered a low shear stress level and is not expected to be detrimental to hMSCs. In previous research, a fluid shear stress of 89 Pa was applied to PC-3 cells while cells traveling through a needle for 6.4 ms, maintaining cell viability at 75%.<sup>30</sup> Human dermal fibroblasts experienced an average shear stress of 190 Pa during the extrusion process with the PU gel showing 65% cell viability.<sup>31</sup>

It is worthy of note that stem cell behaviour and cell fate might be affected by the shear stress. According to previous studies on shear stress regulating hMSCs differentiation, low shear stress ranging from 0.2 to 2 Pa was commonly used.<sup>32–34</sup> The continuous 2 Pa shear stress stimulation for 2 days induces hMSCs toward endothelial cells in the absence of chemical induction.<sup>32</sup> Generally, during the extrusion, experiencing 75 Pa has a limited effect on stem cell fate as it is a short-term process (less than a second), with an undefined long-term effect.

With the excellent printability and non-cytotoxicity of OBD, future work should focus on optimising the cell printing process for practical applications. This includes reducing the time cells spend embedded in a high concentration polymer within the syringe without culture media and using printers that require fewer materials for printing. This could highly



Fig. 6 Bioink printed with cells under red excitation filter (A) and bright field (B). Cells were stained with CellTracker showing orange colour. Scale bar 1 mm.



maintain the cell viability within the cell-laden scaffolds. Improving the binding between the polymer and cells, for example *via* connecting a block of arginine–glycine–aspartate (RGD peptide) to the polymer chain is another way to stabilise cells within the printed scaffold. Providing cell membrane-stabilising agents, such as hydrocortisone, glucose, and glycerol,<sup>8</sup> might also help avoid polymers integrating into the cell membrane and induce cell membrane break through when exposing to shear stress during printing.

## 4. Conclusion

Here, the potential of a non-ionic triblock copolymer as a bioink has been explored. The printability of the polymer was tested across a range of concentrations, with 35% w/w OBD demonstrating great printability, characterised by steady extrusion and a stable structure. The 35% w/w OBD was not cytotoxic to hMSCs, with cell viability exceeding 80% after 3 days of seeding onto the polymer. This viability was improved compared to the commercial thermoresponsive polymer F127. Encapsulating cells within a bioprintable matrix offers exciting possibilities, such as *in situ* printing,<sup>35</sup> personalised structures, and precise control over the printed design.<sup>36</sup> The thermoresponsive properties of this material allow cells to be printed at physiologically relevant temperatures and enable gentle and chemical-free collection of encapsulated or surface-grown cells by liquefying the polymer through temperature reduction.<sup>37</sup> This approach minimises cell stress and preserves cell function and structure, enhancing tissue engineering outcomes.<sup>38</sup> The non-ionic triblock copolymer can also be further functionalised with cell adhesion peptides or other supportive biological motifs, showcasing the versatility of this polymer design. In summary, this thermoresponsive polymer holds significant promise for future bioprinting and tissue engineering applications.

## Data availability

The data supporting this article have been included in the ESI† of this manuscript.

## Conflicts of interest

A. P. C. and T. K. G. report that part of this work regarding the novel chemistry of the materials and their promising thermoresponsive properties have been patented.

## Acknowledgements

A. P. C. acknowledges the Engineering and Physical Sciences Research Council (EPSRC) for the Doctoral Prize Fellowship (EP/M506345/1) as well as the EPSRC Impact Acceleration Grant EP/R511547/1. K. Z. and A. G. acknowledge K. Z.'s PhD scholarship funded by the Australian Government through the Australian Research Council Discovery Project DP200100612.

## References

- M. A. Ward and T. K. Georgiou, Thermoresponsive polymers for biomedical applications, *Polymers*, 2011, **3**(3), 1215–1242, DOI: [10.3390/polym3031215](https://doi.org/10.3390/polym3031215).
- S. Y. Wong, J. M. Pelet and D. Putnam, Polymer systems for gene delivery—Past, present, and future, *Prog. Polym. Sci.*, 2007, **32**(8–9), 799–837, DOI: [10.1016/j.progpolymsci.2007.05.007](https://doi.org/10.1016/j.progpolymsci.2007.05.007).
- N. Yamada, T. Okano, H. Sakai, F. Karikusa, Y. Sawasaki and Y. Sakurai, Thermo-responsive polymeric surfaces; control of attachment and detachment of cultured cells, *Makromol. Chem., Rapid Commun.*, 1990, **11**(11), 571–576, DOI: [10.1002/marc.1990.030111109](https://doi.org/10.1002/marc.1990.030111109).
- E. Ruel-Gariépy and J. C. Leroux, *In situ*-forming hydrogels - Review of temperature-sensitive systems, *Eur. J. Pharm. Biopharm.*, 2004, **58**(2), 409–426, DOI: [10.1016/j.ejpb.2004.03.019](https://doi.org/10.1016/j.ejpb.2004.03.019).
- A. P. Constantinou and T. K. Georgiou, Pre-clinical and clinical applications of thermoreversible hydrogels in biomedical engineering: a review, *Polym. Int.*, 2021, **70**(10), 1433–1448, DOI: [10.1002/pi.6266](https://doi.org/10.1002/pi.6266).
- D. B. Kolesky, R. L. Truby, A. S. Gladman, T. A. Busbee, K. A. Homan and J. A. Lewis, 3D bioprinting of vascularized, heterogeneous cell-laden tissue constructs, *Adv. Mater.*, 2014, **26**(19), 3124–3130, DOI: [10.1002/adma.201305506](https://doi.org/10.1002/adma.201305506).
- M. Kesti, M. Müller and J. Becher, *et al.*, A versatile bioink for three-dimensional printing of cellular scaffolds based on thermally and photo-triggered tandem gelation, *Acta Biomater.*, 2015, **11**(1), 162–172, DOI: [10.1016/j.actbio.2014.09.033](https://doi.org/10.1016/j.actbio.2014.09.033).
- S. F. Khatkhat, S. R. Bhatia and S. C. Roberts, Pluronic F127 as a cell encapsulation material: Utilization of membrane-stabilizing agents, *Tissue Eng.*, 2005, **11**(5–6), 974–983, DOI: [10.1089/ten.2005.11.974](https://doi.org/10.1089/ten.2005.11.974).
- J. W. Nichol, S. T. Koshy, H. Bae, C. M. Hwang, S. Yamanlar and A. Khademhosseini, Cell-laden microengineered gelatin methacrylate hydrogels, *Biomaterials*, 2010, **31**(21), 5536–5544, DOI: [10.1016/j.biomaterials.2010.03.064](https://doi.org/10.1016/j.biomaterials.2010.03.064).
- P. Mozetic, S. M. Giannitelli, M. Gori, M. Trombetta and A. Rainer, Engineering muscle cell alignment through 3D bioprinting, *J. Biomed. Mater. Res., Part A*, 2017, **105**(9), 2582–2588, DOI: [10.1002/jbm.a.36117](https://doi.org/10.1002/jbm.a.36117).
- F. Y. Hsieh and Hsu S. Hui, 3D bioprinting: A new insight into the therapeutic strategy of neural tissue regeneration, *Organogenesis*, 2015, **11**(4), 153–158, DOI: [10.1080/15476278.2015.1123360](https://doi.org/10.1080/15476278.2015.1123360).
- Y. Zhou, Y. Cui and L. Q. Wang, A Dual-sensitive Hydrogel Based on Poly(Lactide-co-Glycolide)-Polyethylene Glycol-Poly(Lactide-co-Glycolide) Block Copolymers for 3D Printing, *Int. J. Bioprint.*, 2021, **7**(3), 140–152, DOI: [10.18063/ijb.v7i3.389](https://doi.org/10.18063/ijb.v7i3.389).
- F. Y. Hsieh, H. H. Lin and Hsu S. Hui, 3D bioprinting of neural stem cell-laden thermoresponsive biodegradable polyurethane hydrogel and potential in central nervous system repair, *Biomaterials*, 2015, **71**, 48–57, DOI: [10.1016/j.biomaterials.2015.08.028](https://doi.org/10.1016/j.biomaterials.2015.08.028).
- C. W. Wong, Y. T. Chen, C. L. Chien, T. Y. Yu, S. P. Rwei and Hsu S. Hui, A simple and efficient feeder-free culture system to up-scale iPSCs on polymeric material surface for use in



- 3D bioprinting, *Mater. Sci. Eng. C*, 2018, **82**(1), 69–79, DOI: [10.1016/j.msec.2017.08.050](https://doi.org/10.1016/j.msec.2017.08.050).
- 15 S. V. Murphy and A. Atala, 3D bioprinting of tissues and organs, *Nat. Biotechnol.*, 2014, **32**(8), 773–785, DOI: [10.1038/nbt.2958](https://doi.org/10.1038/nbt.2958).
- 16 K. Theodoridis, E. Aggelidou and M. E. Manthou, *et al.*, An effective device and method for enhanced cell growth in 3D scaffolds: Investigation of cell seeding and proliferation under static and dynamic conditions, *Mater. Sci. Eng., C*, 2020, **114**(March), 111060, DOI: [10.1016/j.msec.2020.111060](https://doi.org/10.1016/j.msec.2020.111060).
- 17 C. S. Ong, P. Yesantharao and C. Y. Huang, *et al.*, 3D bioprinting using stem cells, *Pediatr. Res.*, 2017, **83**(1–2), 223–231, DOI: [10.1038/pr.2017.252](https://doi.org/10.1038/pr.2017.252).
- 18 A. P. Constantinou, V. Nele and J. J. Douch, *et al.*, Investigation of the Thermogelation of a Promising Biocompatible ABC Triblock Terpolymer and Its Comparison with Pluronic F127, *Macromolecules*, 2022, **55**(5), 1783–1799, DOI: [10.1021/acs.macromol.1c02123](https://doi.org/10.1021/acs.macromol.1c02123).
- 19 A. P. Constantinou, B. Zhan and T. K. Georgiou, Tuning the Gelation of Thermoresponsive Gels Based on Triblock Terpolymers, *Macromolecules*, 2021, **54**(4), 1943–1960, DOI: [10.1021/acs.macromol.0c02533](https://doi.org/10.1021/acs.macromol.0c02533).
- 20 A. Kalarakis, V. Castelletto and C. Chaibundit, *et al.*, Rheology and structures of aqueous gels of triblock(oxethylene/oxbutylene/oxethylene) copolymers with lengthy oxethylene blocks, *Langmuir*, 2001, **17**(14), 4232–4239, DOI: [10.1021/la0101806](https://doi.org/10.1021/la0101806).
- 21 V. Cheng, B. H. Lee, C. Pauken and B. L. Vernon, Poly(N-isopropylacrylamide-co-poly(ethylene glycol))-acrylate simultaneously physically and chemically gelling polymer systems, *J. Appl. Polym. Sci.*, 2007, **106**(2), 1201–1207, DOI: [10.1002/APP.26760](https://doi.org/10.1002/APP.26760).
- 22 G. E. Yu, H. Li and J. P. A. Fairclough, *et al.*, A study of lyotropic mesophases of concentrated solutions of a triblock copolymer of ethylene oxide and 1,2-Butylene oxide, E16B10E16, using rheometry, polarized light microscopy, and small-Angle X-ray scattering, *Langmuir*, 1998, **14**(20), 5782–5788, DOI: [10.1021/la9709453](https://doi.org/10.1021/la9709453).
- 23 C. O'Connell, J. Ren and L. Pope, *et al.*, Characterizing Bioinks for Extrusion Bioprinting: Printability and Rheology, *3D Bioprinting*, 2020, pp. 111–133, DOI: [10.1007/978-1-0716-0520-2\\_7](https://doi.org/10.1007/978-1-0716-0520-2_7).
- 24 J. Schindelin, I. Arganda-Carreras and E. Frise, *et al.*, Fiji: An open-source platform for biological-image analysis, *Nat. Methods*, 2012, **9**(7), 676–682, DOI: [10.1038/nmeth.2019](https://doi.org/10.1038/nmeth.2019).
- 25 M. Müller, J. Becher, M. Schnabelrauch and M. Zenobi-Wong, Nanostructured Pluronic hydrogels as bioinks for 3D bioprinting, *Biofabrication*, 2015, **7**(3), 035006, DOI: [10.1088/1758-5090/7/3/035006](https://doi.org/10.1088/1758-5090/7/3/035006).
- 26 K. Suman, S. Sourav and Y. M. Joshi, Rheological signatures of gel-glass transition and a revised phase diagram of an aqueous triblock copolymer solution of Pluronic F127, *Phys. Fluids*, 2021, **33**(7), 073610, DOI: [10.1063/5.0057090](https://doi.org/10.1063/5.0057090).
- 27 J.-M. Brunet-Maheu, J. C. Fernandes, C. A. V. de Lacerda, Q. Shi, M. Benderdour and P. Lavigne, Pluronic F-127 as a cell carrier for bone tissue engineering, *J. Biomater. Appl.*, 2009, **24**(3), 275–287, DOI: [10.1177/0885328208096534](https://doi.org/10.1177/0885328208096534).
- 28 S. L. Bellis, Advantages of RGD peptides for directing cell association with biomaterials, *Biomaterials*, 2011, **32**(18), 4205–4210, DOI: [10.1016/j.biomaterials.2011.02.029](https://doi.org/10.1016/j.biomaterials.2011.02.029).
- 29 T. Billiet, E. Gevaert, T. De Schryver, M. Cornelissen and P. Dubruel, The 3D printing of gelatin methacrylamide cell-laden tissue-engineered constructs with high cell viability, *Biomaterials*, 2014, **35**(1), 49–62, DOI: [10.1016/j.biomaterials.2013.09.078](https://doi.org/10.1016/j.biomaterials.2013.09.078).
- 30 J. M. Barnes, J. T. Nauseef and M. D. Henry, Resistance to Fluid Shear Stress Is a Conserved Biophysical Property of Malignant Cells, *PLoS One*, 2012, **7**(12), e50973, DOI: [10.1371/journal.pone.0050973](https://doi.org/10.1371/journal.pone.0050973).
- 31 L. Ho and Hsu S. Hui, Cell reprogramming by 3D bioprinting of human fibroblasts in polyurethane hydrogel for fabrication of neural-like constructs, *Acta Biomater.*, 2018, **70**, 57–70, DOI: [10.1016/j.actbio.2018.01.044](https://doi.org/10.1016/j.actbio.2018.01.044).
- 32 L. Yuan, N. Sakamoto, G. Song and M. Sato, High-level shear stress stimulates endothelial differentiation and VEGF secretion by human mesenchymal stem cells, *Cell. Mol. Bioeng.*, 2013, **6**(2), 220–229, DOI: [10.1007/s12195-013-0275-x](https://doi.org/10.1007/s12195-013-0275-x).
- 33 Dong J. De, Y. Q. Gu and C. M. Li, *et al.*, Response of mesenchymal stem cells to shear stress in tissue-engineered vascular grafts, *Acta Pharmacol. Sin.*, 2009, **30**(5), 530–536, DOI: [10.1038/aps.2009.40](https://doi.org/10.1038/aps.2009.40).
- 34 D. H. Kim, S. J. Heo, S. H. Kim, J. W. Shin, S. H. Park and J. W. Shin, Shear stress magnitude is critical in regulating the differentiation of mesenchymal stem cells even with endothelial growth medium, *Biotechnol. Lett.*, 2011, **33**(12), 2351–2359, DOI: [10.1007/s10529-011-0706-5](https://doi.org/10.1007/s10529-011-0706-5).
- 35 C. Di Bella, S. Duchi and C. D. O'Connell, *et al.*, In situ handheld three-dimensional bioprinting for cartilage regeneration, *J. Tissue Eng. Regen. Med.*, 2018, **12**(3), 611–621, DOI: [10.1002/TERM.2476](https://doi.org/10.1002/TERM.2476).
- 36 A. V. Do, B. Khorsand, S. M. Geary and A. K. Salem, 3D Printing of Scaffolds for Tissue Regeneration Applications, *Adv. Healthcare Mater.*, 2015, **4**(12), 1742–1762, DOI: [10.1002/ADHM.201500168](https://doi.org/10.1002/ADHM.201500168).
- 37 T. Takezawa, Y. Mori and K. Yoshizato, Cell Culture on a Thermo-Responsive Polymer Surface, *Nat. Biotechnol.*, 1990, **8**(9), 854–856, DOI: [10.1038/nbt0990-854](https://doi.org/10.1038/nbt0990-854).
- 38 F. Doberenz, K. Zeng, C. Willems, K. Zhang and T. Groth, Thermoresponsive polymers and their biomedical application in tissue engineering – a review, *J. Mater. Chem. B*, 2020, **8**(4), 607–628, DOI: [10.1039/C9TB02052G](https://doi.org/10.1039/C9TB02052G).

