

REVIEW

[View Article Online](#)
[View Journal](#) | [View Issue](#)Cite this: *Mater. Horiz.*, 2025,
12, 4998Polyelectrolyte complex-based materials for
separations: progress, challenges
and opportunitiesJiaying Li,^a Lijie Li,^{ab} Hestie A. Brink,^b Giulia Allegri^a and
Saskia Lindhoud^{id} ^{★a}

Polyelectrolyte complex (PEC) based materials could provide a sustainable alternative to conventional materials, especially for separation applications. However, reproducible production remains a challenge due to the many parameters influencing the polyelectrolyte complexation process, eventually affecting the properties and performance of the final material. Here, we provide an overview of how different parameters affect polyelectrolyte complexation and discuss promising PEC-based materials for separation applications, *i.e.*, porous membranes, functional and barrier coatings, adhesives, saloplastics, and extraction media. Additionally, we highlight the challenges and opportunities and discuss what is needed to get to the next level. We envision that collaboration between experimentalists and theoreticians can leverage experimental datasets with accurate descriptions of all the parameters for multiscale modelling, machine learning and artificial intelligence approaches that can be used to design PEC materials and predict their properties.

Received 16th December 2024,
Accepted 20th March 2025

DOI: 10.1039/d4mh01840k

rsc.li/materials-horizons

Wider impact

Modern materials based on petroleum-derived polymers are mainly processed by organic solvents and are difficult to recycle. Polyelectrolyte complex-based materials that are processable in water can be recycled and have demonstrated self-healing properties. These properties make them interesting candidates for renewable materials. Although polyelectrolyte complexes have been studied for over a century, many aspects are not fully understood, and it is challenging to make accurate theoretical descriptions for these systems. Reproducible production of polyelectrolyte complexes has been hampered by batch-to-batch variation of the starting material and the way of mixing solutions containing oppositely charged polyelectrolytes affects the properties of the polyelectrolyte complex. In addition, there are many parameters influencing polyelectrolyte complexation, *e.g.*, the chemistry of the polymers, their length, their mixing ratio, the salt concentration, pH, and temperature resulting in a multidimensional phase space which is difficult to systematically study. To overcome these challenges and elevate our understanding of these interesting materials to a higher level, open-access datasets with metadata that accurately describe the system, *i.e.*, how the sample was prepared, are needed. These datasets can be used to improve the theory and will allow for multiscale modelling and AI approaches to make predictions of material properties.

1. Introduction

For more than a century, researchers have been interested in polyelectrolyte complexation, a phenomenon in which mixing aqueous solutions of oppositely charged polyelectrolytes causes the system phase to separate into a dense phase and a dilute phase. As early as 1911, Tiebackx was the first to report this

phenomenon when he mixed aqueous solutions of gelatin and gum arabic.¹ In the late 1920s, Bungenberg de Jong and Kruyt further described this formation of liquid–liquid phase separation. They named this type of phase separation “complex coacervation”.² From then on, researchers have been studying biopolymer complex coacervation systematically, mainly focussing on proteins and polysaccharides.^{3,4} When synthetic polyelectrolytes became available, the formation of solid-like complexes was first described by Fuoss and Sadek in the 1940s.⁵ Later in 1957, the first theory on complex coacervation was reported by Overbeek and Voorn.⁶ The potential of polyelectrolyte complexes (PECs) as a material was soon recognised by Michaels in 1965.⁷ In this paper, he envisioned the possible applications of PECs based on their unique properties,

^a Department of Molecules and Materials, Faculty of Science and Technology, MESA+ Institute for Nanotechnology, University of Twente, P.O. Box 217, 7500 AE Enschede, The Netherlands. E-mail: s.lindhoud@utwente.nl

^b Department of Membrane Science and Technology, Faculty of Science and Technology, MESA+ Institute for Nanotechnology, University of Twente, P.O. Box 217, 7500 AE Enschede, The Netherlands

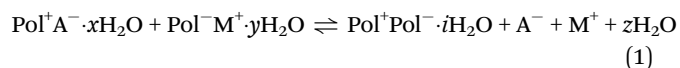


including in membranes for medical and electrical applications, plastic composites, conductive coatings, and environmental sensors.

Now, about 60 years later, single polyelectrolytes are commercially available for water treatment as flocculants, for household products as rheology modifiers, and for diapers as a superabsorbent material.⁸ However, the commercial use of PEC-based materials is rare. If these materials are so promising, why are PEC-based materials not on the market yet? Especially because PECs offer several advantages, including low toxicity⁹ and high recyclability potential,^{10–12} therefore these materials could play a pivotal role in the transition to more sustainable materials. Furthermore, PECs are processed using water as a solvent, which reduces the use of harmful chemicals and minimises the environmental impact of the manufacturing process. Additionally, PEC materials have self-healing properties,^{8,12–14} which can extend the lifespan of products and reduce the need for frequent replacements.

In this review, we address this question by examining current research works on PECs, summarising the potential and limitations of their materials, and providing possible solutions to overcome these challenges.

Polyelectrolytes are polymers with ionisable functional groups. These functional groups are mainly amine-based for polycations, and sulphate/carboxylic group based for polyanions. Bediako *et al.* illustrated the chemical structures of the most commonly used polyelectrolytes categorised as natural, semi-synthetic, and synthetic.¹⁵ To maintain electroneutrality in solution, the polyelectrolytes are accompanied by oppositely charged counterions. Upon mixing, the entropic gain from releasing the counterions is the main driving force for the formation of PECs, as shown in eqn (1):^{8,16}



where Pol^+ represents the polycation chain associated with the A^- counterion and Pol^- represents the polyanion chain

associated with the M^+ counterion. After complexation, Pol^+Pol^- represents the intrinsic polycation–polyanion pair while the counterions M^+ and A^- are released. This equation directly shows the importance of ionic strength as a parameter to control the PEC formation, *e.g.*, above a certain salt concentration the entropic gain of complex formation ceases to exist.

In addition, considering eqn (1), the mixing ratio between the polyelectrolytes is important. Depending on the salt concentration and composition of the PEC system, different phases will be encountered as shown in Fig. 1a.¹⁷ If one starts with a polycation solution at low salt concentration upon the addition of polyanions, first positively charged soluble complexes will form, and subsequently, at a certain mixing ratio, macroscopic phase separation occurs. In theory, maximal complexation is observed at charge stoichiometry, the ratio at which the number of positively and negatively charged monomers of the polyelectrolytes are equal ($\text{Pol}^+:\text{Pol}^- = 1$). Further increasing the amount of polyanions will dissolve the complex and negatively charged soluble complexes will be present in the solution. At low salt concentration in general, solid-like PECs are formed, and increasing the salt concentration may lead to the formation of liquid-like complex coacervates. At high salt concentration the PEC phase dissolves, and a one phase system is obtained. Apart from salt, pH is an important tuning parameter when the system contains weak polyelectrolytes, *i.e.*, polyelectrolytes whose degree of ionisation is dependent on the pH.¹⁸ Weak polycations are fully charged at low pH and uncharged at high pH, and the weak polyanions are fully charged at high pH and uncharged at low pH as shown in Fig. 1b. To make it more complex, when both weak polyanions and polycations are present in the system these molecules can affect each other's dissociation behaviour.^{19–21}

Focusing on sustainability, most of the discussed polyelectrolytes in this review are water-soluble and/or can be processed using water as the main solvent. These polyelectrolytes are hygroscopic and for many of their materials, their properties depend on the presence of water, *i.e.*, many PEC-based materials

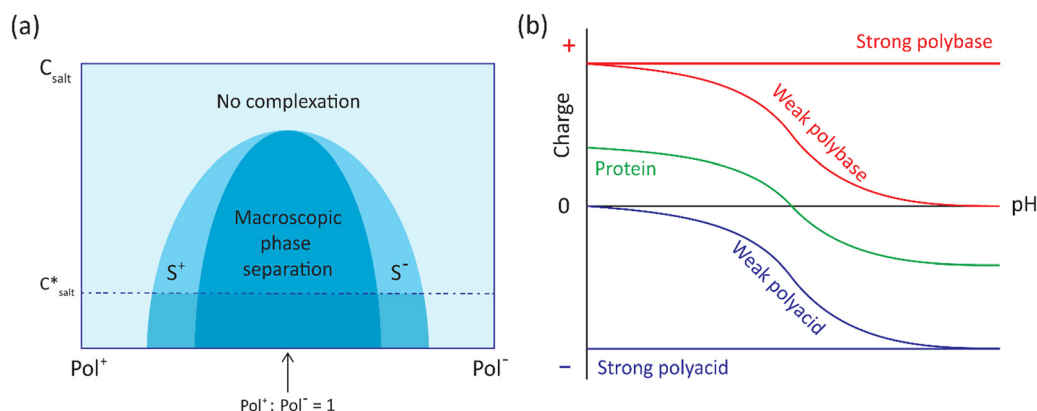
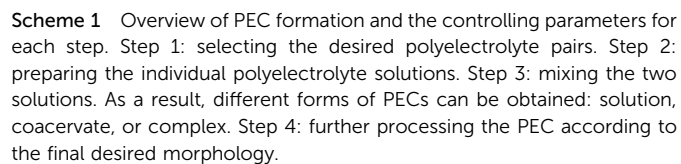


Fig. 1 (a) Phase diagram of polyelectrolyte complexation between the oppositely charged polycation (Pol^+) and polyanion (Pol^-) as a function of charge stoichiometry and salt concentration (C_{salt}). S^+ and S^- represent soluble PECs. C_{salt}^* is the critical salt concentration, beyond which liquid-like coacervate can form or at high salt concentration, there is no complexation. Reproduced with permission.¹⁷ Copyright 2002, American Chemical Society. Below C_{salt}^* solid-like PECs form. (b) Charge density of polyelectrolytes and proteins *versus* pH as a function. Reproduced with permission.¹⁸ Copyright 2019, Elsevier.



The most studied method to construct PEC materials, however, is not included in Scheme 1. One reason why PECs themselves became less popular to study is the development of layer-by-layer (LbL) assembly of polyelectrolytes. Instead of step 3 of mixing, LbL directly fabricates polyelectrolyte multilayers (PEMs) by sequential deposition of polycations and polyanions. First proposed by Decher in the 1990s,^{42–45} the development of LbL accelerated the development of PEMs, and many different types of polyelectrolytes and substrates can be used, and different parameters can be easily tuned to build this confined structure. Many reviews have elaborated on the development of LbL focusing on both theories and their potential applications with a particular interest in biomedical fields, such as drug delivery and biosensing,^{46–53} and membranes.^{54–58} As shown in Fig. 2, LbL has dominated the study of polyelectrolyte complexation since its discovery, in most of the applications suggested by Michaels. Although research-wise LbL has been developed in all directions and has shown great potential, commercially, there are very limited products. Some commercial PEM-based nanofiltration membranes are available by NX filtration and Pentair.⁵⁹ The industrialization of PEMs is limited by the time-consuming steps and in all cases, a substrate is required, which in the case of membranes is not always produced *via* a sustainable process. This is why we need to rethink how to utilize bulk PECs as Michaels proposed, using water as the solvent.



Total publications: 7312 (PEC: 38.4%, LbL 61.6%)

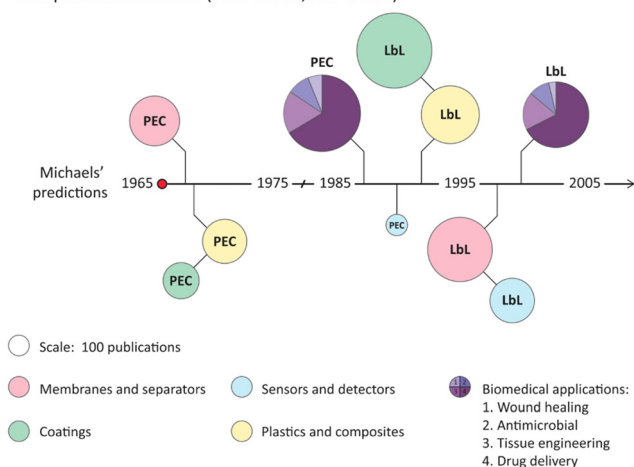


Fig. 2 Number of publications of PECs and LbL in 5 main applications. The keywords used for searching in Scopus were “polyelectrolyte AND “layer by layer” AND application” for LbL and “polyelectrolyte complex*”-AND application AND NOT “layer by layer”” for PECs. For example, when searching membranes, the key words are “polyelectrolyte AND “layer by layer” AND membrane*” and “polyelectrolyte complex*” AND membrane* AND NOT “layer by layer””, correspondingly. The search was performed within the article title, abstract, and keywords. To be noted, there could still be overlaps for different fields or between LbL and PECs. The year states the first paper that was published in the field. The numbers were obtained on 26th August 2024.

Apart from layer-by-layer assemblies, in biomedical applications, PECs, especially complex coacervates, are a well-studied topic as highlighted by several reviews.^{60–63} These applications also make use of the responsiveness of PECs to changes in the (local) environment. In many cases, the PECs have dimensions in the nanometre range. In this review, we will focus on non-biomedical and non-LbL prepared macroscopic PEC-based materials, produced starting from (1) homogenous polyelectrolyte solutions, (2) polyelectrolyte complexes, and (3) considering both the complex phase and the dilute phase. We will focus on porous membranes, functional barrier coatings, adhesives, saloplastics, and aqueous-based extraction media. Most of these PEC-based materials were proposed by Michaels. We will discuss the state-of-the-art and what challenges *e.g.*, reproducibility issues, need to be overcome, to bring these materials to higher technology readiness levels. The goal is to set all practical criteria and show how far we are from replacing current polymer materials. In addition, to bridge the gap between theoretical predictions and experiments, standards need to be set to build reliable datasets. These datasets could then also be used for multiscale modelling, machine learning, and artificial intelligence (AI) and could be used to make predictions of material properties.

2. From starting solution(s) to material

Depending on the material to be produced, different starting solutions need to be prepared as shown in Scheme 1. In this section, we will first focus on materials that are prepared starting with homogenous solutions containing both oppositely

charged polyelectrolytes. Then we will discuss materials for which the starting point is either a homogenous solution dispersion of polyelectrolyte complexes or a complex coacervate. Followed by a discussion on materials of which a polyelectrolyte complex is the starting point. We conclude with aqueous two-phase systems in which both the complex phase and the supernatant phase play a role.

Homogenous solutions containing oppositely charged polyelectrolytes can be obtained at a high salt concentration or, when one of the polyelectrolytes has a pH-dependent charge, mixing the polyelectrolytes at a pH at which the weak polyelectrolyte is uncharged. By decreasing the salt concentration or changing the pH, polyelectrolyte complexation can be induced. This principle can be used to produce porous membranes and functional coatings. Currently, for the preparation of these materials organic solvents are being used and replacing these solvents by water would make the production process more environmentally friendly.

2.1 PEC membranes for water treatment

Because of their unique transport properties and ability to selectively separate compounds polymer membranes have been widely used in agriculture,⁶⁴ medicine,⁶⁵ food,⁶⁶ environment, and energy⁶⁷ industries.^{68,69} Currently, non-solvent induced phase separation (NIPS) is the predominant method for polymeric membrane production because of its simplicity and efficiency, and it allows for preparing defect-free membranes for a large industrial scale with broad performance.^{70,71} The NIPS process includes the preparation of dope solutions and phase separation in a non-solvent bath, typically water. To get the dope or casting solutions, organic solvents that are mixable with water, such as *N*-methyl pyrrolidone (NMP), dimethyl formamide (DMF), or dimethyl acetamide (DMAc) are commonly used to dissolve the polymers that are water insoluble. When these polymer solutions are cast on a support plate and submerged in water, the organic solvent, water exchange, and the polymeric membranes form. NIPS relies on organic solvents which are unsustainable and harmful to the environment.⁷² In 2018, the European Council imposed a restriction on the use of NMP, so it is urgent to find other solutions to prepare membranes in a more sustainable way.⁷³

The unique properties of PECs such as non-solubility in common organic solvents, high compatibility, and stability under environmental conditions make PECs suitable materials for membrane preparation.^{74,75} Michaels *et al.* discussed the possibility of utilizing PECs to prepare porous membranes. However, solid PECs are difficult to process, and one research direction is to tune the composition of polyelectrolyte solutions to obtain processable PECs.⁷ Successful examples include acid-protection and complexation-sulfation to produce processable solid-like PECs. These solid-like PECs were first made by adding acid or sulfated groups and then dissolving them in suitable aqueous solutions to obtain PEC solutions, followed by a casting and drying process to obtain PEC membranes.^{76–79} The acid blending method adds excess acid into weak polyelectrolyte solution to depress the ionic complexation in low-



concentration polyelectrolyte solutions.^{80,81} Some reviews have provided comprehensive discussions on the preparation and applications of different PEC-based membranes.^{74,82,83} They showed great performance for applications in pervaporation, water treatment, and electrodialysis, *etc.*^{74,82} However, these PEC-based membranes were prepared *via* a drying process and need a supporting substrate and there is a lack of control over the membrane structures and performance.

Recently, it was shown that by changing the solubility of solutions containing oppositely charged polyelectrolytes, free-standing PEC-based membranes can be produced *via* the NIPS-inspired aqueous phase separation (APS) method. The preparation process is similar to NIPS, a homogenous casting solution is prepared first, and then a solid membrane is obtained *via* phase separation in a coagulation bath. Unlike NIPS using non-solvents to induce phase separation, the phase separation of APS is induced *via* the change in pH/salinity of the bath solutions. In Scheme 1, the starting point is homogeneous polyelectrolyte solutions obtained after mixing polycation and polyanion solutions, for membrane formation the viscosity of these solutions is very important for the structure and membrane performance. In addition, the formation of PEC membranes, *i.e.*, phase separation, can also be controlled by tuning the parameters in the coagulation bath. The development of APS opened a new way to prepare free-standing PEC-based membranes, and the APS-produced membranes demonstrated tunable structures and performance. Below we will discuss the APS approach for pH change-induced and salinity change-induced membrane formation, the advantages of this approach, and the challenges to overcome to produce membranes *via* this sustainable method.

2.1.1 pH change-induced APS. For pH change-induced APS homogenous casting solutions are obtained by mixing oppositely charged polyelectrolytes of which one is weak and one is strong. Considering Fig. 1b, these homogenous solutions can be obtained by mixing a weak polyanion and strong polycation at low pH or a weak polycation and strong polyanion at high pH.⁸⁴ Under these conditions, polyelectrolyte complexation will not occur and homogenous casting solutions can be obtained.

The APS approach was first shown to work for the preparation of single polyelectrolyte membranes based on the pH-responsive solubility of weak polyelectrolytes.^{85–88} Poly(4-vinyl pyridine) (P4VP) was dissolved at a low pH where it is charged

and soluble ($\text{pH} < 4$), then its solution was cast as a thin film and switched to a high pH where P4VP is uncharged and insoluble. This process led to the phase separation and solidification of P4VP. Both symmetric and asymmetric membranes with controlled structures were obtained by tuning the pH difference between the casting solution and coagulation bath. The APS approach was also used to prepare membranes using copolymer polystyrene-*alt*-maleic acid (PSaMA). Like P4VP, the solubility of this polymer depends on the pH of the casting solution, but for PSaMA the polymer is soluble in alkaline water and can be precipitated in an acidic water bath. By adding different types of weak acids in the coagulation bath, membranes with dense separation layers were obtained and possessed an average rejection $> 92\%$ towards the diverse range of micropollutants. These two examples show that by changing the solubility of the polymer through switching the pH from low to high or *vice versa*, similar to the conventional NIPS approach, free-standing porous polymeric membranes can be obtained.

In 2020, the APS approach was successfully used to prepare PEC membranes for oppositely charged polyelectrolyte pairs. Baig *et al.* first prepared PEC membranes based on the complexation of strong polycation poly(sodium 4-styrene sulfonate) (PSS) and weak polyanion poly(allylamine hydrochloride) (PAH).^{89,90} As shown in Fig. 3a, homogenous casting solutions were prepared at a high pH ($\text{pH} \sim 14$) to discharge PAH and the PEC membranes were obtained at a low pH ($\text{pH} \sim 1$). Like NIPS, APS demonstrated great control over the PEC membrane structure and separation performance by tuning the composition of the casting solutions such as the polyelectrolyte molecular weight, concentration, and mixing ratio. Besides, the addition of sodium chloride in the coagulation bath also tuned the membrane structure, which is a new tuning parameter for membrane preparation compared to the NIPS process. The APS-based PAH–PSS membranes demonstrated desirable separation performance from microfiltration to nanofiltration. One disadvantage was that the formation of the PAH–PSS casting solution needed the addition of enough sodium hydroxide to make PAH uncharged. For another polyelectrolyte pair, polyethyleneimine (PEI) and PSS, a homogenous casting solution could be obtained by directly mixing PEI and PSS without further pH adjustment, and the APS was induced through a milder

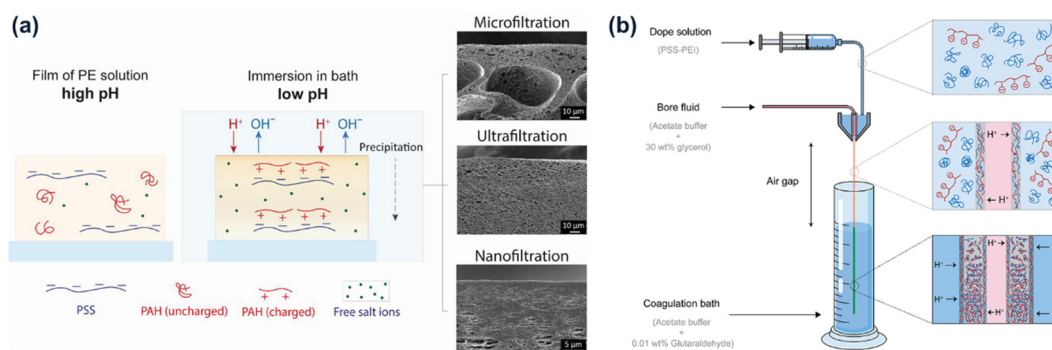
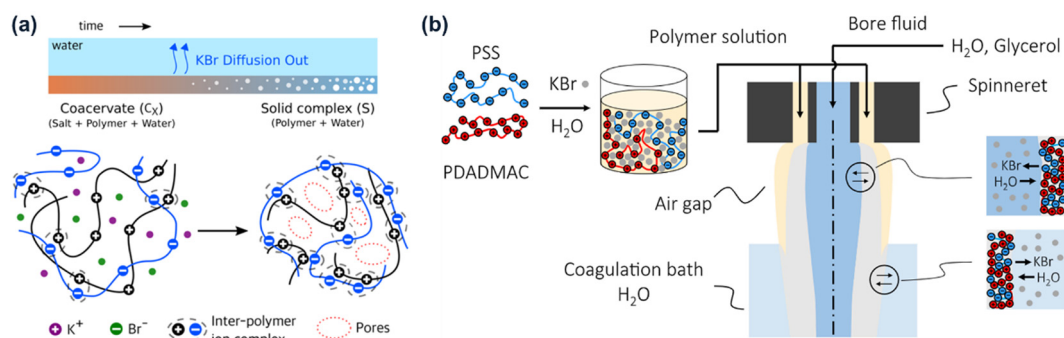


Fig. 3 Schematic illustration of PEC-based (a) flat membranes (reproduced with permission.⁸⁹ Copyright 2020, Wiley) and (b) hollow-fiber membranes (reproduced with permission.⁹⁴ Copyright 2022, American Chemical Society) *via* pH change-induced APS.



2.1.2 Salinity change-induced APS. The pH change-induced APS is used to prepare PEC membranes when there is a weak polyelectrolyte present. However, strong polyelectrolytes have a charge that is independent of the pH of the system. It is known that above a certain salt concentration the charges of the polyelectrolytes are screened and homogenous solutions containing oppositely charged polyelectrolytes can be obtained.^{96,97} When the solution with high salt concentration is cast and then immersed in a bath, *i.e.* pure water, due to the large difference in salt concentration, the excess ions can immediately diffuse into the water. As the concentration of salt ions decreases, the driving force for complexation increases and then leads to the formation of solid PEC membranes.

It should be noticed that for salinity change-induced APS, the choice of salts is important. In the work of preparing QVP-C2-PSS membranes, it was found that the solid complex remained stable in many salt solutions such as NaCl, KCl, CaCl₂, and MgCl₂, and only KBr caused high swelling of the



Mater. Horiz., 2025, 12, 4998-5030 | 5003

complex. For the PDADMAC–PSS system, both NaCl and KBr can be used for dissolving the complex. Therefore, the suitable types of salts are limited based on the polyelectrolyte pairs, and in our opinion a thorough understanding of polyelectrolyte counterion interactions is desirable for the production of APS membranes. Salts that have relatively high solvation-free energy are more efficient in breaking the PEC ion pairs and thus are more suitable as the salt for APS.⁹⁸

2.1.3 Challenges and opportunities. Free-standing PEC membranes based on different polyelectrolyte pairs have been successfully prepared *via* pH change or salinity change-induced APS. Table 1 is a summary of all the types of PEC-based membranes produced *via* the APS approach. Compared to the previous methods of using PECs to prepare membranes, the APS method makes it possible to prepare membranes using bulk PECs *via* a single-step phase separation and without using support membranes. The APS is inspired by the conventional NIPS method for the production of membranes while it eliminates the use of organic solvents. It also demonstrates advantages similar to NIPS. The membrane structure and performance can be well-tuned by the composition of polyelectrolyte solutions and the coagulation bath, including the polyelectrolyte concentration, molecular weight, mixing ratio, pH, and salinity. Besides, chemical cross-linking can be done in the coagulation bath with the formation of PECs simultaneously. Using the APS approach, membranes with microfiltration, ultrafiltration, and nanofiltration performance can be prepared. In addition, it is possible to prepare both flat sheet and hollow-fiber membranes. Therefore, the APS showed great potential for preparing membranes with controlled structure and separation performance, which expands the ways of utilizing PECs as membrane materials.

However, there are also problems existing in the APS method. Here, we summarise the main challenges and also possible improvements from the following aspects, tracing back to Scheme 1 in the introduction:

(a) Choice of polyelectrolytes: for now, the APS approach has only been applied to limited polyelectrolyte pairs. A suitable

polyelectrolyte pair is vital for the successful preparation of PEC membranes. The pH change-induced APS needs weak polyelectrolytes that are pH-responsive. The salt change-induced APS can use strong polyelectrolytes to prepare PEC membranes while suitable salt and high salinity are needed. The properties of the polyelectrolytes, such as molecular weight, acid dissociation constant (pK_a) value, and film-forming property, should be considered before preparing the membranes.

(b) Preparation of polyelectrolyte solutions: the APS approach starts from a homogenous polyelectrolyte solution. The solution properties, such as polymer concentrations, viscosity, pH, and ionic strength, directly affect the final structure and performance of the membranes. To obtain membranes with high mechanical strength, it is better to use higher polymer concentrations. However, this leads to an increase in the viscosity, and high viscosity can affect the processability of the solutions. A balance between the polymer concentration and solution viscosity is very important. A possible solution is to increase the temperature to obtain a higher polyelectrolyte concentration with low viscosity. Because increasing temperature can improve the mobility of polymer chains and thus decrease the intrinsic viscosity.

(c) Preparation of membranes: the membranes are formed from the phase separation of polyelectrolyte solutions in the coagulation bath, and their structure can be tuned by the composition of the coagulation bath, such as pH and ionic strength. The driving force from the pH or salinity between the casting solution and the coagulation bath has to be high enough to make stable PEC membranes. Besides, additives/crosslinkers can be added to the bath to make membranes with denser structures.

(d) Reproducibility: this problem is especially important for the membranes prepared from natural-based polyelectrolytes.⁹⁵ This is because the membranes are formed through the complexation of oppositely charged polyelectrolytes. Different batches or sources of polyelectrolytes showed differences in charge density which would therefore affect the membrane formation process. The charge density of polyelectrolytes also

Table 1 Summary of different types of PEC-based membranes produced *via* the APS approach

Type of PECs	Condition of APS	Studied parameters	Types of membranes	Ref.
PAH–PSS	pH ~ 14 to pH ~ 1	PE MW, concentration, and mixing ratio; salinity of bath; crosslinker (GA)	Microfiltration Ultrafiltration Nanofiltration	89, 90
PEI–PSS	pH ~ 11 to pH ~ 4	PE MW and mixing ratio; temperature of casting solution; pH and concentration of acetate buffer bath; crosslinker (GA)	Microfiltration Ultrafiltration Nanofiltration	91, 92
PEI–PSS hollow fiber membranes	pH ~ 11 to pH ~ 4	Glycerol concentration; pH and concentration of acetate buffer bath	Microfiltration Ultrafiltration	94
PDADMAC–PAA	pH ~ 1 or 3 to pH ~ 4	Casting solution pH; PE MW and mixing ratio; pH and salinity of bath	Microfiltration	93
CS–CMC	pH ~ 1 to pH ~ 5	pH and concentration of acetate buffer bath	Microfiltration	95
QVP–C2–PSS	Salinity change (KBr)	Salt concentration in the coacervate	Microfiltration Ultrafiltration	98
PDADMAC–PSS	Salinity change (NaCl)	PE MW and concentration; salinity of bath	Nanofiltration	99
PDADMAC–PSS	Salinity change (KBr)	PE mixing ratio	Ultrafiltration Nanofiltration	100
PDADMAC–PSS tubular membranes	Salinity change (KBr)	Glycerol concentration	Nanofiltration	101



influences the solution's viscosity, which is an important factor that affects the phase separation process.^{89,103} A full characterization of the polyelectrolytes, such as molecular weight and charge density, will be important before preparing the membranes.

(e) Applications: the NIPS-produced membranes have been used in various fields, as the APS method of preparing PEC membranes is still new, APS-produced membranes have only demonstrated potential in water treatment fields, and more research needs to be done to extend the applications. For example, developing more natural polyelectrolyte membranes for biomedical applications.

The APS approach also opens new opportunities for membrane functionalisation. APS-produced membranes can be used as support membranes for coating PEMs to enhance the separation performance of the membranes.¹⁰⁴ A film formed by the one-step APS can also act as a coating to functionalise hollow fiber membranes.¹⁰⁵ Furthermore, since water is the solvent, it is possible to introduce biocatalysts into the APS-produced membranes. Different from the LbL or coating method, in APS, the biocatalyst can be directly added into the polyelectrolyte solutions. After the phase separation process, membranes with catalytic functions are obtained. So far, biocatalytic PAH-PSS and PEI-PSS membranes functionalised with lysozyme have been successfully prepared *via* the pH change-induced APS.^{103,106} In addition, alkaline phosphatase was immobilized on the PDADMAC-PSS hollow fiber membranes through the salinity change-induced APS.¹⁰² For appropriate characterization of the enzymatic activity in these membranes, having a methodology to determine the concentration of enzymes that are incorporated in these membranes is desirable. Nevertheless, membranes with added biocatalytic functions are not only interesting for applications in water purification, but these materials could also be interesting for biomedical purposes.

Overall, the development of the APS approach reveals that bulk PEC materials can be well used in controllable membrane preparation. The APS method shows great control of the structure and performance of membranes, just like the traditional NIPS. The APS-produced membranes demonstrated separation ability from microfiltration to nanofiltration, which are capable of separating various species in water treatment. Besides, the APS approach can be used to functionalise the membranes.

2.2 PEC functional coatings

For coating applications, one could start with a homogenous solution, with the viscous complex coacervate phase or a polyelectrolyte complex dispersion, so any starting point in Scheme 1. Coatings are defined as functional layers deposited on a substrate, serving specific purposes like protection or decoration.¹⁰⁷ The coatings are usually a continuous film and have a small fraction of the thickness when compared to the bulk substrate.¹⁰⁸ To reduce the use of volatile organic components (VOCs), waterborne coatings such as polyurethanes and acrylic polymers have been developed.¹⁰⁹ Polyelectrolytes, especially the water-soluble ones, are powerful tools for surface modifications. For the deposition of PEC coatings, the most commonly used method is LbL. The electrostatic interaction

between two oppositely charged polyelectrolytes is utilized to build up layers by repeatedly dipping, spraying, *etc.* Richardson *et al.* summarised a wide range of methods to prepare such LbL films.⁵² The LbL assembly method is straightforward and versatile since it is not limited by forms of interactions and the type of substrates. As discussed in the introduction, the boost of the LbL development has helped to realize the use of PEMs. Michaels proposed PECs as conductive/antistatic coatings according to the charged nature of polyelectrolytes.⁷ He suggested that these specific functionalities could be achieved by adding conductive components or playing with the stoichiometric ratio. There are a few studies regarding this specific area, including PEDOT/PSS for conductive coatings,¹¹⁰ copper ion-crosslinked sodium alginate-graphene oxide for antistatic and antibacterial fabric coatings,¹¹¹ PEI/GO conductive coatings,¹¹² polyelectrolytes and ITO nanoparticles for conductive paper coatings,¹¹³ chitosan/sodium phytate/TiO₂-CuO nanoparticle composites for flame-retardant and antistatic wood coatings,¹¹⁴ carboxymethyl cellulose/surfactant complexes for antistatic paper coatings,¹¹⁵ and chitosan/hyaluronic acid complexes with electric responsive behaviour.¹¹⁶ Most of these coatings were prepared *via* LbL, and LbL as a powerful method can not only easily construct polyelectrolyte-polyelectrolyte/surfactant complexes but also immobilize all types of functional nanocomponents.

Although LbL has shown great potential in various research fields, the major hindrance for its industrial application is its laborious preparation. With the conventional immersion method, coating one bilayer (a layer of positively charged polyelectrolyte and a layer of negatively charged polyelectrolyte) typically takes 45 min and is mostly done manually.¹¹⁷ To upgrade the method, researchers have tried to narrow the time for each step or by using an automated robot.^{118,119} Other approaches have also been developed, such as automated spraying,¹²⁰ however, a true single-step deposition method is still lacking. Here, we focus on PEC coatings prepared in limited steps or one step which are usually beyond the nm thickness range.

One approach is building from PEC particles, similar to conventional film formation of coatings where a polymer dispersion is first deposited, followed by coalescence.¹²¹ Wang *et al.* developed a coating strategy by spraying poly(L-lysine) (PLL) and hyaluronan (HA) complex followed by humidity curing.¹²² By directly mixing PLL and HA, nanoparticles of PEC were formed which were stable for more than 24 h. These particles were then sprayed and cured on glass at different time lengths and relative humidity (RH) to achieve homogeneous films. These coatings showed potential to be used as extracellular matrix (ECM) membranes. Basu *et al.* prepared CS and CMC fiber-like particles by high-speed mixing.¹²³ The barrier properties of coated paperboard substrates showed significant improvements against both water and oil after heat treatment. Later, Chi and Catchmark further improved the barrier properties by adding crystalline nanocellulose, showing that the method can also allow for adding other reinforcements.¹²⁴ Chi *et al.* also prepared such complex particles utilizing cationic and anionic starches. The key finding was that high MW and charge density were essential to form a dense network for



Table 2 Summary of PEC particle-based functional coatings and their preparation methods

Type of complex	Preparation methods	Functionality and performance	Ref.
PLL and HA	PE ⁺ :PE ⁻ (v/v) = 0.5:1.0 to 2.0:1.0 Mixing: strong stirring Spraying: 0–30 min using an ultrasonic spray device, glass substrates Annealing: under 60 or 100% RH conditions at 25 °C for 0–1 h	Successful incorporation of 7-hydroxycoumarin (7-HC), FITC labelled bovine serum albumin (BSA-FITC), DNAs, and VEGF	122
CS and CMC	PE ⁺ :PE ⁻ (wt%) = 1:1 Mixing: blended at 25 000 rpm for 5–60 min at pH 3.0 and 4.5 Dip coating: only one side in contact, paperboard substrates Annealing: dried in an oven at 140 °C for 10–15 min	Oil and water barrier properties at room temperature and 80 °C, also resistance against water vapor (transmission rate: 60 g mm d ⁻¹ m ⁻²), toluene, <i>n</i> -heptane, salt solutions	123
Cationic and anionic starches	PE ⁺ :PE ⁻ (wt%) = 1:1 Mixing: shear homogenization at 1500 rpm for 5 min Dip coating: only one side in contact, paperboard substrates Annealing: dried in an oven at 150 °C for 10 min	Improved mechanical properties (18% increase in tensile strength and 21% increase in Young's modulus), and excellent barrier against water vapor (40% lowered), grease (kit number of 12), and oil penetration (Cobb ₆₀ : 1.85 g m ⁻²)	125

barrier properties.¹²⁵ Some controlling parameters and potential applications are summarised in Table 2.

Another promising deposition method of PEC coating is starting from coacervate. Grunlan's group was the first to study PEC coacervate coatings and their potential functionalities, starting with PEI and PAA (Fig. 5a).¹²⁶ Both pH and salt concentration were tuned, and the thickness can be controlled by the casting rod. Stronger complexation was achieved by post-treatment in a buffer bath. This specific combination of polyelectrolytes has been proven to work as an excellent gas barrier as PEM,¹²⁷ here, this single-step coating also showed excellent oxygen barrier properties. Later, the same group applied this

method to different combinations of PEC:PDADMAC and PAA.¹²⁸ Instead of starting from a coacervate, a solution was prepared by keeping PAA uncharged at pH 2 and using a pH switch discussed in the introduction (Scheme 1). The substrate was dip-coated with this solution, then dried, and a buffer bath was used to initiate the complexation (Fig. 5b). The resulted films again exhibited excellent oxygen barrier properties. To further enhance the barrier properties, clays, such as kaolinite, can also be investigated as an additive.¹²⁹ Edible PECs of CS and pectin (PT) were also successfully prepared while this method showed a good oxygen barrier for fruit.¹³⁰ The application of PDADMAC and PAA complex coating reduced bacterial adhesion preventing bacterial fouling.¹³¹ Another main focus of Grunlan's group is flame retardant coatings. Polyamine and polyphosphate are used to form the layer, following a similar method starting with deposition followed by using a buffer bath. More details are discussed in their review focusing on flame retardant functionality and some examples are given in Table 3.¹³²

More recently, a single-step evaporation-induced film formation of PEC was first proposed by Pietsch *et al.*¹³⁶ then developed by Li *et al.* As shown in Fig. 5c, a homogeneous solution with polyanions can be obtained at high pH when keeping PEI uncharged.¹³³ The unique point is that the pH change was induced by an evaporative base ammonia, in this way, complexation can be triggered during drying. This method has proven to work for PEI and PAA or PSS.^{135,137} Since PEI was kept uncharged, homogeneous solutions with different charge ratios could be prepared. The viscosity was influenced by both MW and polymer concentrations. The film thickness could be controlled easily by the casting rod. Again, the combination of PEI and PAA showed excellent oxygen barrier properties.¹³⁵ The experimental details and performance of some mentioned works are summarised in Table 3.

2.2.1 Challenges and opportunities. We discussed how PEC-based coatings can be obtained starting from different starting points: dispersed PEC particles, complex coacervate, and solution. Among all properties, serving as barrier coatings is the most promising application. The main reason could be that the dense ionic-crosslinking acts as a good barrier, meanwhile, thicker films are also beneficial for good coverage and better barrier properties. In his review, Hubbe thoroughly

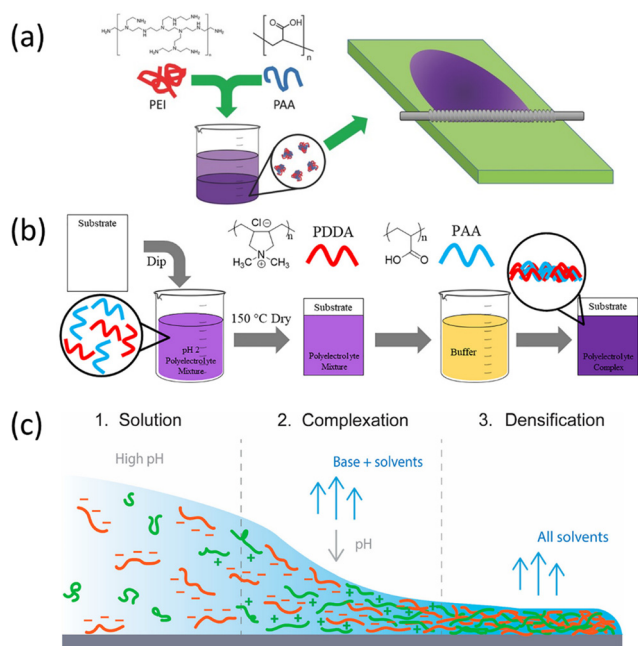


Fig. 5 Examples of PEC coacervate/solution-deposited coatings: (a) PEI/PAA coacervate oxygen barrier coatings. Reproduced with permission.¹²⁶ Copyright 2017, Wiley-VCH. (b) PDADMAC/PAA solution oxygen barrier coatings, reproduced with permission.¹²⁸ Copyright 2018, American Chemical Society. (c) Film formation mechanism of evaporation-induced complexation coatings. Green: PEI, orange: polyanion, reproduced with permission.¹³³ Copyright 2024, American Chemical Society.



Table 3 Summary of PEC coacervate/solution-prepared functional coatings and their preparation methods

Type of complex	Preparation methods	Functionality and performance	Ref.
PEI and PAA	PE ⁺ :PE ⁻ (wt%) = 1:1 Mixing: stirring at pH 8 and different salt concentrations (0–1.5 M NaCl), then heated for 2 h at 70 °C Casting: using a hand-drawn rod on poly(ethylene terephthalate) (PET) Annealing: immersed into citric acid/citrate buffer solutions at pH 2, 4, and 6 in combination with humidity and heating treatments	Lowest oxygen transmission rate (OTR) 0.08 cm ³ (m ² d atm) ⁻¹	126
CS and PT	PE ⁺ :PE ⁻ (wt%) = 1:3 Mixing: rolling for at least 12 h Dip coating: PET immersed in PEC for 5 min and then dried at 70 °C for 5 min and 150 °C for 1 h in an oven or dried with blowing Annealing: cured with pH 5 200 mM CA buffer for 5 min, followed by water rinsing and the same drying procedures	Prevention of browning of apple slices for 3 h, banana was kept green for 1 week, blow-dried coating has an OTR of 0.291 cm ³ m ⁻² day ⁻¹ atm ⁻¹ and an oxygen permeability of 1.02 × 10 ⁻¹⁸ cm ³ cm cm ⁻² s ⁻¹ Pa ⁻¹	130
PDADMAC and PAA	PE ⁺ :PE ⁻ (wt%) = 1:3 Mixing: magnetically stirred overnight at pH 2 Dip coating: polyester fabric immersed in PEC for 5 min and dried for 3 h at 70 °C Annealing: immersed in 200 × 10 ⁻³ M citric acid at pH 3 or 5 for 20 min, washed and dried at 70 °C overnight	Removal of >95% of deposited <i>Staphylococcus aureus</i> after rinsing with water	131
PEI and poly(phosphate sodium salt) (PSP)	PE ⁺ :PE ⁻ (wt%) = 1:2 Mixing: PEI was poured into PSP, both with a pH of 7 Dip coating: cotton fabric immersed in PEC for 30 min then hung to dry in a 70 °C oven for 3 h	52.7% reduction in total heat release	134
PEI and PAA	PE ⁺ :PE ⁻ = 4:1 to 1:3 Mixing: NH ₃ was first added to PAA, then PAA-NH ₃ was added into PEI solution. The mixture was mixed vigorously for 30 min. Casting: using a casting machine with different thicknesses of Meyer rods on biaxially orientated polypropylene (BOPP) substrates	Lowest oxygen permeability below 4 cm ³ m ⁻² day ⁻¹ atm ⁻¹ (<0.002 barrer)	135

discussed PEC coatings as a barrier, especially for packaging.¹³⁸ Here, we also summarise the key challenges and possible future improvements from the following aspects, tracing back to Scheme 1 in the introduction:

(a) Choice of polyelectrolytes: to achieve specific functionality, it could be controlled by the intrinsic properties of individual polyelectrolytes, such as polar PAA as the gas-barrier¹²⁶ and PSP as the fire retardant.¹³⁹ For the film formation, high MW is usually desired to have enough entanglements plus mechanical stability. This also applies to the charge density that normally high charge density is desired for good complexation. Biobased polyelectrolytes are promising, especially for edible coatings for food preservation.¹³⁰ The design of one-step PEC coatings can use LbL coatings as a guide since the same functionality can be reached with a different approach.

(b) Preparation of PECs: the ratio between polycation and polyanion is crucial for the final properties of PEC coatings. As shown in Tables 2 and 3, however, it is often not studied as the charge molar ratio. Depending on the final form of PECs, different parameters should be considered. For PEC particles, to achieve uniform particle size, strong stirring is needed. In principle, there are wide choices of polyelectrolytes for forming these particles. The difficulty remains in the coalescence of the particles as defects can form during the film formation and the formation of a continuous film is not achieved. For complex coacervates, again there are many choices of polyelectrolytes. The salt and ratio between the polyelectrolytes can be tuned to optimize the viscosity of the complex coacervates. Compared to particles, complex coacervates are more processible, and compared to solutions, they are one step closer to complexation.

The problem could be the unknown concentration and composition of the formed complex coacervates. The last form of starting PEC is high-concentration solutions where the complexation was avoided by keeping the weak polyelectrolyte uncharged. In this way, the ratio can be easily adjusted, and the final composition is known. However, the choices of polyelectrolytes are limited since certain pK_a values of the weak polyelectrolyte are required.

(c) Deposition methods: despite the form of PECs, viscosity is the most important parameter. For conventional water-borne coatings, viscosity modifiers can be used to adjust the viscosity, which are not yet studied in PEC coating formulations. The viscosity of PECs is mostly controlled by the polyelectrolyte concentration. According to the viscosity and the desired end product, different methods can be chosen, such as spraying, dip coating, or casting. Among these methods, casting is the most convenient, and thickness can be easily controlled.

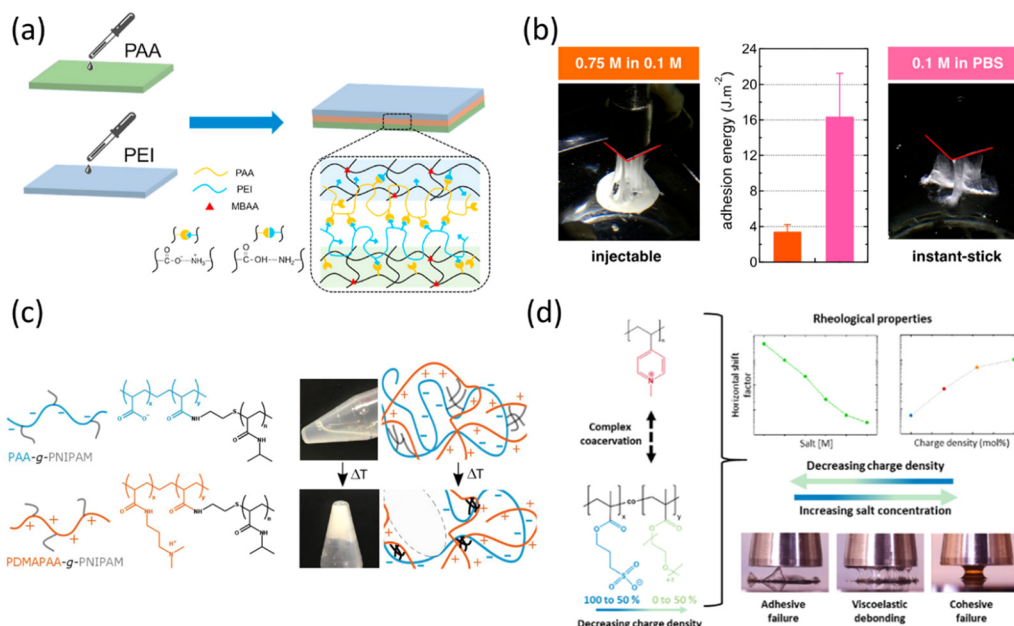
(d) Post-treatments: as summarised in Tables 2 and 3, post-treatments including a buffer bath, heating, humidity treatment, or chemical crosslinking can allow rearrangements, enhance complexation, or provide extra strength. The resulting films usually show less swelling and a smoother surface.

(e) Additives: additives could be used to improve the properties or induce functionality, for example, nanoclays were added to enhance the gas and water barrier properties.¹²⁹ The compatibility between the additives and PECs is key so that no phase separation should occur during the preparation. The challenge is to achieve an even distribution within the PEC matrix.

(f) Durability/stability: for functional coatings, it is important to study the stability of the coating. With time or damage, delamination or defects that compromise the functionality



2.3.1 Synthetic PEC-based adhesives. Zhang *et al.* proposed a method to adhere two hydrogels, polyvinyl pyrrolidone (PVP)-PAA and poly(vinyl alcohol)-polyacrylamide (PVA-PAAm), by separately coating them with PAA and PEI solutions as shown in Fig. 6a.¹⁷⁴ After removing the excess solutions, the coated hydrogels were pressed together to allow for contact, and the strong electrostatic interaction between PAA and PEI led to the



This journal is © The Royal Society of Chemistry 2025

adhesion. This method was further improved by introducing Fe^{3+} to the PAA solution first. The competition between cations, Fe^{3+} and NH_4^+ from PEI, slowed down the PEI–PAA complexation, which resulted in better and repeatable adhesion. This straightforward coating method was the first to utilize two commercially available polyelectrolytes and demonstrated the possibility of PEC for wet adhesion.

Held together by ionic crosslinking, salt naturally can be utilized as a trigger for tuning the adhesion strength. Vahdati *et al.* studied the coacervate formation of poly(2-acrylamido-2-methylpropanesulfonic acid) (PAMPS) and poly(*N,N*-(dimethylamino)propyl methacrylamide) (PMADAP).¹⁷⁵ As shown in Fig. 6b, the complex coacervate remained injectable at 0.75 M NaCl, while at 0.1 M, the complex coacervate was closer to the gel-point, thus exhibited instant stickiness. One crucial parameter to achieve underwater adhesion found by this study was the low molecular weight. With the degrees of polymerization (DP) close to 100, a system with suitable water content, unentangled chains, and appropriate sol–gel transition can be obtained which fits the requirements for biomedical applications.

Other parameters can also be optimized to trigger wet adhesion, such as pH and temperature. Kamperman's group has been focusing on thermoresponsive poly(*N*-isopropylacrylamide) (PNIPAM) based on coacervates. Dompé *et al.* grafted both PAA and poly(dimethylaminopropyl acrylamide) (PDAMAPAA) with PNIPAM chains. As shown in Fig. 6c, a temperature trigger was embedded so that above the lower critical solution temperature (LCST), a liquid-to-gel transition can be achieved.¹⁷⁶ When the temperature was raised to 50 °C, PNIPAM chains formed physical crosslinked domains which strengthened the coacervate. Later, a follow-up study focused on the salt effect on this system.¹⁷⁸ When the salt concentration of the environment was lowered to 0.1 M NaCl, salt ions diffused out from the coacervate, initiating a stronger electrostatic interaction. As a result, better adhesion could be achieved at 20 °C when compared to using a temperature switch. Van Hees *et al.* synthesized PNIPAM-*b*-poly(acrylic acid)-*b*-PNIPAM and poly(2-(dimethylamino)ethyl methacrylate) (PDMAEMA) to form complex coacervates.¹⁷⁹ A similar result was found, that increasing the temperature and lowering the salt content can both alter the viscoelastic behaviour of the complex coacervate. This study further stressed the importance of determining the optimal PNIPAM content in order to balance all properties.

Focusing on the effect of charge density on adhesion strength, van Westerveld *et al.* conducted two similar studies by keeping the polycation poly(*N*-methyl-4-vinylpyridinium iodide) (QP4VP) the same, while tuning the hydrophobicity of the polyanions: poly(3-sulfopropyl methacrylate) (PSPMA) partially substituted with oligo([ethylene glycol]methyl ether methacrylate) (OEGMA) units P(SPMA-*x*-co-OEGMA)_{*y*}¹⁷⁷ and P(BSPMA-*x*-SPMA).¹⁸⁰ Taking the P(SPMA-*x*-co-OEGMA)_{*y*} and QP4VP complex coacervate system as an example (Fig. 6d), diluting the charge of one of the polyelectrolytes can also be used as a parameter to tune the viscoelastic behaviour of the coacervate. This strategy is more feasible than the salt switch since for biomedical applications, the salt concentration is typically low.

2.3.2 Bio-based PEC adhesives. Adhesion, both dry and wet, achieved by using bio-based polyelectrolytes is highly desired due to their biocompatibility and biodegradability. Here, we showcase some examples that utilized at least one bio-based polyelectrolyte with or without modifications. Shao and Stewart synthesized poly-phosphate and poly-aminated gelatin to mimic the mechanism of adhesion of marine worms.¹⁸¹ Together with the presence of divalent cations Ca^{2+} or Mg^{2+} , the initiation of the wet adhesion of this coacervate can be triggered by salt, pH, or temperature (Fig. 7a). When measured at 37 °C and pH 7.4, the lap shear strength increased when increasing the divalent cation content. In this work, low molecular weight, non-gelling gelatin (MW 3–5 kDa) was used, which again showed the importance of using low molecular weight.

Another example is using CMC and polyvinyl amine (PVAm) colloidal complexes to improve wet paper strength where wet adhesion is important.¹⁸² As shown in Fig. 7b, the wet adhesion was enhanced when more amines were presented, which was contradictory to the results under dry conditions. The formation of the complex did not benefit the adhesion properties when compared to only the use of PVAm. The advantage of using this complex is that a higher amount could be deposited than a linear-soluble polymer. This study also emphasized that the adhesion of PVAm may be mainly from primary amines, instead of quaternary amines since PDADMAC did not show wet adhesion properties.

Some other studies have investigated lignosulfonate as the polyanion, which is a waste product derived from wood during the sulphite pulping processes.¹⁸⁶ Sodium lignosulfonate ($\text{L-SO}_3\text{Na}$) was used to form complexes with three different polycations: poly(allylamine) (PAH), PDADMAC, and ϵ -poly-L-lysine (ϵ -PL).¹⁸⁷ The importance of the substrates was emphasized since these complexes showed much better adhesion on polar substrates like metal and wood than non-polar plastic polypropylene. Among the three combinations, $\text{L-SO}_3\text{Na}/\epsilon$ -PL complexes are the most promising since both polyelectrolytes are biomass-derived. Another example is the mixing of lignosulfonate (LS) and a polyamidoamine-epichlorohydrin (PAE-Cl) solution.¹⁸³ Instant underwater adhesion could be achieved and the self-curing ability of PAE further improved the wet adhesion during curing (Fig. 7c). Again, in this work, better adhesion was achieved with hydrophilic substrates than hydrophobic substrate PTFE.

Chitosan is a derivative of abundant crustacean sources, such as crab and shrimp shells, which contain positively charged amine groups.¹⁸⁸ Waite *et al.* designed a coacervate system using catechol-functionalised PAA and bis(trifluoromethane)sulfonamide (TF_2N^-) modified quaternised chitosan.¹⁸⁹ TF_2N^- groups were used to increase the solubility of chitosan in DMSO. These two polyelectrolytes were premixed in DMSO and then applied on different substrates in water. The phase inversion was then activated by water–DMSO solvent exchange, charging the carboxylic groups, eventually leading to wet adhesion. This is a unique work, which used solvent exchange for stimuli, however, one review has pointed out that the use of DMSO may be harmful to the bio-organisms.¹⁶⁸ In another work, Li *et al.* utilized CS or quaternary ammonium salt of chitosan (QCS) to form hydrogels with sodium alginate (SA), as



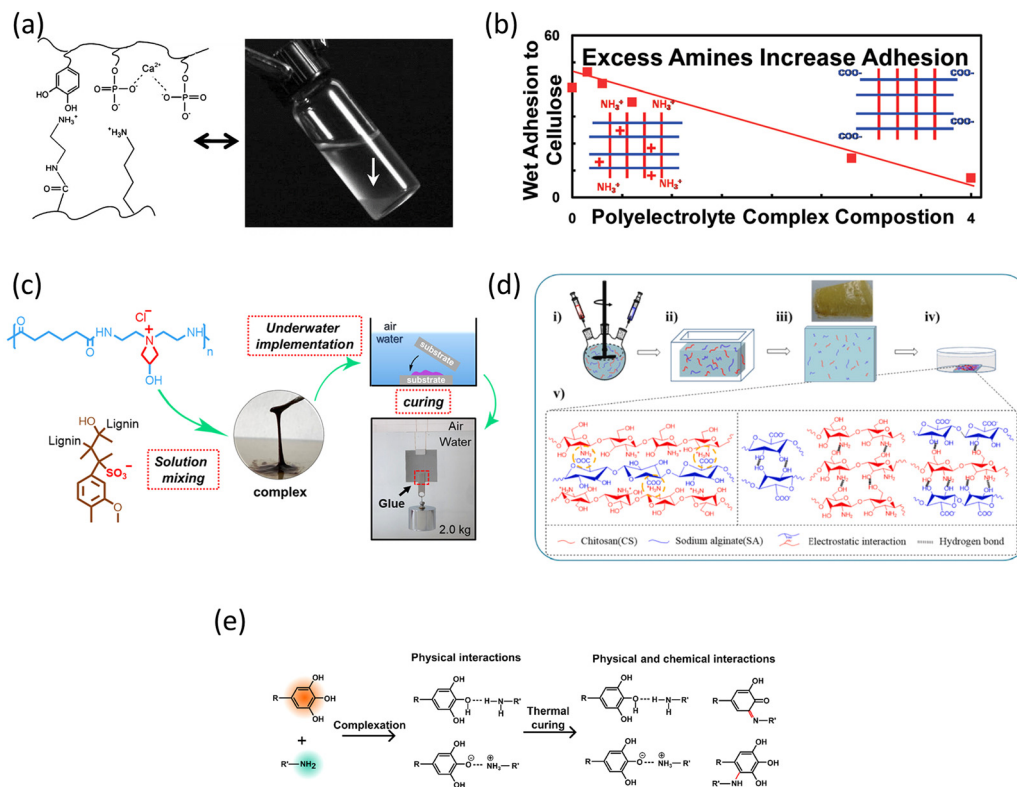


Fig. 7 Examples of adhesive coacervates prepared by one or two biobased polyelectrolytes: (a) coacervate formation of polyphosphate-polyaminated gelatin-divalent cation, reproduced with permission.¹⁸¹ Copyright 2010, Wiley-VCH. (b) PVAm and CMC strengthened wet cellulose, reproduced with permission.¹⁸² Copyright 2007, American Chemical Society. (c) LS and PAE-Cl underwater adhesive, reproduced with permission.¹⁸³ Copyright 2019, American Chemical Society. (d) CS and SA hydrogel, reproduced with permission.¹⁸⁴ Copyright 2022, Elsevier. (e) Thermal-cured CS and TA coacervate. Reproduced with permission.¹⁸⁵ Copyright 2024, American Chemical Society.

shown in Fig. 7d.¹⁸⁴ High molecular weight polyelectrolytes (approx. 10^5 Da) were used which contributed to the high strength due to both hydrogen bonding and electrostatic interaction. The obtained CS/SA hydrogels showed excellent adhesion to various biological tissues. The addition of Ag^+ further improved the strength and the antibacterial performance. Similarly, Wei *et al.* fabricated an injectable hydrogel based on QCS and carboxymethylcellulose sodium (CMCNa).¹⁹⁰ The obtained hydrogel coating exhibited excellent interfacial properties such as superspreading on various substrates underwater, maintaining interfacial toughness, and improving lubrication. Another fully biobased system was proposed by Galland *et al.* where they used CS and hyaluronic acid (HA).¹⁹¹ After examining the salt-induced changes, one big advantage of this complex coacervate was that no large trigger was required to induce adhesion. The last example is using CS and tannic acid (TA) with a curing step (Fig. 7e).¹⁸⁵ Besides the adhesion strength, this mild temperature treatment improved both water resistance and long-time durability underwater. In general, the usage of chitosan or quaternised chitosan as the polycation can also offer other functionalities such as antibacterial properties, which was also studied in some of these mentioned works.^{57,191}

2.3.3 Adhesive by two positively charged polyelectrolytes.

In most cases, complex coacervates are formed by two oppositely charged polyelectrolytes undergoing a liquid-liquid phase separation driven by the electrostatic attraction and an

entropy gain from releasing the counterions.¹⁹² Here, a unique complex coacervation with two positively charged polyelectrolytes has been reported by Kim *et al.*¹⁹³ The recombinant mussel foot protein-1 (Rmfp-1, M(AKPSYPPTYK)₁₂) and poly(2-(trimethylamino)ethyl methacrylate) (MADQUAT) were used to form the complex coacervate. This complex coacervation is mainly dominated by the cation- π interaction between trimethyl ammonium and phenol groups as shown in Fig. 8, which was strong enough to overcome the electrostatic repulsion. More studies have been focused on this cation- π interaction because of its stability in saline environments, where coacervation based on electrostatic interaction would fail due to screening of the charges.^{194,195}

2.3.4 Challenges and opportunities. Despite all these promising studies, the development of underwater adhesives is still in its early stage, and far from industrial applications.¹⁷⁰ In this section, we will discuss the current challenges and possible future directions in developing PECs as underwater adhesives.

(a) Choice of polyelectrolytes: as discussed in previous sections, the use of polyelectrolytes can be categorized in two main directions: design synthetic polyelectrolytes to mimic biological wet adhesion or utilize biobased polyelectrolytes, such as polysaccharides. For synthetic polyelectrolytes, the introduction of hydrophobic components can help the first stage of adhesion, which is removing the interfacial water layer. It is thus important to balance the ratio between hydrophilic



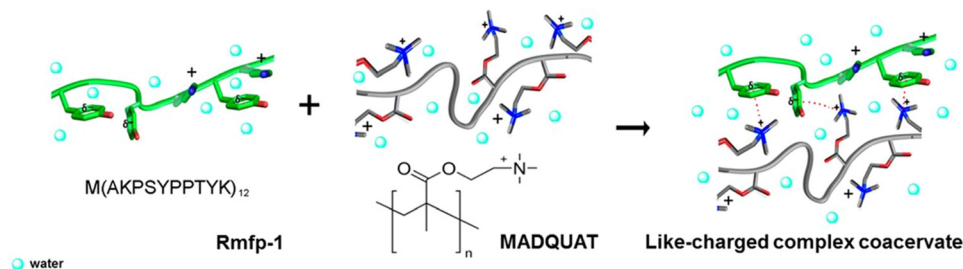


Fig. 8 Coacervation of Rmfp-1 and MADQUAT based on cation- π interaction. Reproduced with permission.¹⁹³ Copyright 2016, National Academy of Sciences.

and hydrophobic parts.¹⁶⁸ For biobased PECs, Lu *et al.* pointed out that there is a trade-off between biocompatibility and mechanical strength when using natural adhesives.¹⁷³ They tend to have short life spans thus it is important to study the self-healing properties.¹⁷² One important parameter is the MW. Using lower MW polyelectrolytes is easier for processing due to lower viscosities, however, sacrifices the final mechanical stability due to lack of entanglements. One suggestion was to combine small and large MW polyelectrolytes or use polyelectrolytes with large PDIs.¹⁷¹ In addition, other weak interactions are often combined with electrostatic interactions to work synergically towards better adhesion, such as hydrophobic interaction, hydrogen bonding, and metal-ligand coordination.¹⁷¹

(b) Phase inversion: the phase inversion of PECs is a powerful tool to transform from injectable/flowable liquid to strong solid-like adhesive. In Table 4, the stimuli conditions for selected examples are summarised. The most important factors studied are temperature, pH, and ionic strength. Here, Vahdati *et al.* concluded that high salt concentration and high-temperature switches are not practical for biomedical uses since the tolerance for both is normally low. Another factor is that a stable water content should be achieved for consistent adhesion performance. The final water content of the adhesives also determines whether they are suitable for hydrophilic or hydrophobic surfaces.¹⁷¹ Charge stoichiometry is also studied in this field. Around a 1 : 1 charge ratio is optimal probably due to the maximum stability at neutral charge.

For future developments, Cui *et al.* proposed to construct phase diagrams showing pH, ionic strength, polyelectrolyte concentration *etc.* as guidelines for future designs.¹⁹⁶ Natural adhesives produced by organisms are dynamic materials that enable the adhesion process to proceed at different lengths, and time scales, thus it is vital to design and study the adhesion with changing parameters.^{170,196} Furthermore, the environmental impact should also be taken into account, for example, when using solvent-exchange as the trigger which may be harmful to the environment.¹⁸⁹

(c) Stability: due to the nature of electrostatic interactions, it is difficult for PEC-based adhesives to remain stable in sea water applications.¹⁶⁸ Also, some of the PECs are sensitive to both ionic strength and pH, which makes it hard for them to sustain dynamic flows, such as blood.¹⁶⁹ In general, there lacks an injectable adhesive which cures at body temperature and adheres to biological tissues with specificity to the target

surfaces under complex body fluidic conditions.¹⁶⁸ The long-time adhesion was only investigated in a few studies (Table 4), which should also be considered.

(d) Characterization techniques: one general limitation emphasized by Narayanan *et al.* is the lack of standardized measurement methods.¹⁶⁹ As summarised in Table 5, the adhesion strength was measured using various methods and under different experimental conditions, for example, some of them were not conducted underwater. This makes the parallel comparison difficult. Despite the difference in measurements, Vahdati *et al.* set a standard for robust adhesion strength which should be larger than 100 J m^{-2} , however, most of the examples could not reach this value yet.¹⁷¹

By the hydrophilic nature of PECs, most of these adhesives work better on hydrophilic substrates. They also mentioned the roughness of the substrate was less studied where most of work used smooth surfaces. In real applications, surface roughness may also play a role.¹⁶⁹ Li *et al.* emphasized the importance of developing characterization techniques to understand the chemical compositions down to molecular levels for biological adhesives, which also holds true for PEC adhesives.¹⁷²

In summary, complex coacervation is proven to be a vital step for marine organisms to achieve wet adhesion. There are still many unknowns in fully understanding the biological and applicational processes of the wet adhesion thus it is still at the research stage. In the future, it is important to further study the coacervation mechanism and design the structures of the polyelectrolytes. Proteins and biobased polyelectrolytes should be the focus due to their biocompatibility and biodegradability. These adhesives can be promising materials for biomedical applications such as surgical use, wound dressing, and multi-functional bioelectronics.^{168,197,198}

2.4 Saloplastics as separators and functionalised materials

In this section, we will examine the processing of PECs into dense plastic materials. As illustrated in Scheme 1, the preparation of a PEC is relatively straightforward, involving the mixing of oppositely charged polyelectrolytes at low salt concentrations, which results in the aggregation and precipitation of a solid-like complex.¹⁹⁹ Historically, the processing of these complexes has presented significant challenges due to their inherent brittleness in the dry state and their insolubility in water and most solvents.²⁰⁰ This changed in 2009 when the Schlenoff group highlighted the importance of saltwater in the



Table 4 Summary of selected PEC adhesives with different dominant interactions and their stimuli for adhesion

Type of complex	Type of interactions	Adhesion conditions	Stability over time	Ref.
PEI and PAA (with Fe^{3+})	Electrostatic interaction, hydrogen bonding, metal–ligand coordination	Curing time: 24 h Temperature: 25 °C pH: adjusted to 3 for samples with Fe^{3+}	Intact after 3-day soaking in water	174
PAMPS and PMADAP	Electrostatic interaction	$\text{PE}^+ : \text{PE}^- = 53.2\text{--}46.8$ mol% Curing conditions: 1 h Temperature: 20–25 °C pH: 7 Salt switch: 0.75 M to 0.1 M NaCl; 0.1 M NaCl to PBS (physiological condition) Water content: 56 wt% (0.1 M sample)	N/A	175
PDMAPAA-g-PNIPAM and PAA-g-PNIPAM	Electrostatic interaction, hydrophobic interaction	$\text{PE}^+ : \text{PE}^- = 1 : 1$ Curing conditions: until a fixed thickness was reached, then 1 h contact pH: 7 Temperature switch: from 20 °C to 50 °C when keeping the NaCl concentration at 0.75 M Salt switch: from 0.75 M to 0.1 M NaCl when keeping the temperature at 20 °C Combined salt and temperature switch: 0.75 M to 0.1 M NaCl and 20 °C to 50 °C Water content: 83.1–92.9 wt%	N/A	176, 178
P(SPMA _x -co-OEGMA _y) and QP4VP	Electrostatic interaction, hydrophobic interaction	$\text{PE}^+ : \text{PE}^- = 1 : 1$ Curing conditions: 1 min contact Salt concentrations: 0.00 M vs. 0.5 M NaCl Water content: 51–84 wt%	N/A	177
Polyphosphate–gelatin–divalent cation complexes	Electrostatic interaction	$\text{PE}^+ : \text{PE}^- = 0.6$ Curing conditions: fully submerged in water for ≈ 24 h Temperature: 10, 20, and 37 °C pH: 7.4	N/A	181
PVAm and CMC	Electrostatic interaction	$\text{PE}^+ : \text{PE}^-$ (wt%) = 0.25 to 1 Curing conditions: equilibrated at 23 °C and 50% humidity for 24 h pH: 4 to 9 Salt concentration: 0.01 M NaCl	N/A	182
PAE-Cl and LS	Electrostatic interaction, hydrophilic stabilization	$\text{PE}^+ : \text{PE}^-$ (wt%) = 1.25 : 1 Curing conditions: 48 h conditioning vs. instant in different soaking solutions Temperature: 50, 80, and 100 °C pH: 3, 5, water, 8, 9, 10, and 11 Salt concentrations: water, 0.2, 0.5, 0.8, and 1.0 M NaCl	Stable up to 30 days	183
CS/QCS and SA	Electrostatic interaction, hydrogen bonding	$\text{PE}^+ : \text{PE}^-$ (wt%) = 1 : 0.4 to 1 : 1.6 Curing conditions: 1 min contact Water content: 67.6–95.3 wt% for CS/SA and 50.3–83.9 wt% for HAC/SA	N/A	57
CS and HA	Electrostatic interaction, hydrogen bonding	$\text{PE}^+ : \text{PE}^-$ (wt%) = 1 : 1.1 Temperature: 37 °C pH: 5 Salt concentrations: supernatant, 0.1, 0.2, 0.3, 0.4 and 0.45 M NaCl Water content: > 85 wt%	Maintains 80% of its adhesion strength after 24 h of immersion in PBS	191
CS and TA	Electrostatic interaction, hydrogen bonding, and cation– π interaction	$\text{PE}^+ : \text{PE}^-$ (wt%) = 1 : 0.25 to 1 : 10 Curing conditions: cured in the oven at 70 °C for 2 h Temperature: room temperature	Immersion in water for 1 day to 2 months (3.5 MPa maintained after 2 months)	185
MADQUAT and Rmfp-1	Cation– π interaction	MADQUAT : Rmfp-1 = 1 : 1 (stoichiometry of tyrosine/trimethylammonium units) pH: 3 Curing time: at least 10 min	N/A	193

processing of PECs, and introduced the term “saloplastics” to describe the resulting materials. The presence of salt facilitates the doping of the complex,^{200,201} effectively disrupting the ionic crosslinks between the anionic and cationic groups, which allows for the rearrangement of polymer chains. Additionally, hydration of the PEC is essential, as water serves as a plasticiser for these materials.¹⁰ Furthermore, saloplastic processing typically occurs at elevated temperatures, as a temperature/doping

equivalence¹⁰ exists. This relationship indicates that plasticization is enhanced at higher temperatures, thereby accelerating the processing of saloplastic materials.

By controlling the PEC composition and studying their behaviour in different environments, it should be possible to functionalise the properties and morphologies of saloplastic materials for a diverse range of applications. The selection of polyelectrolyte pairs and their respective molecular weights



Mater. Horiz., 2025, **12**, 4998–5030 | 5013

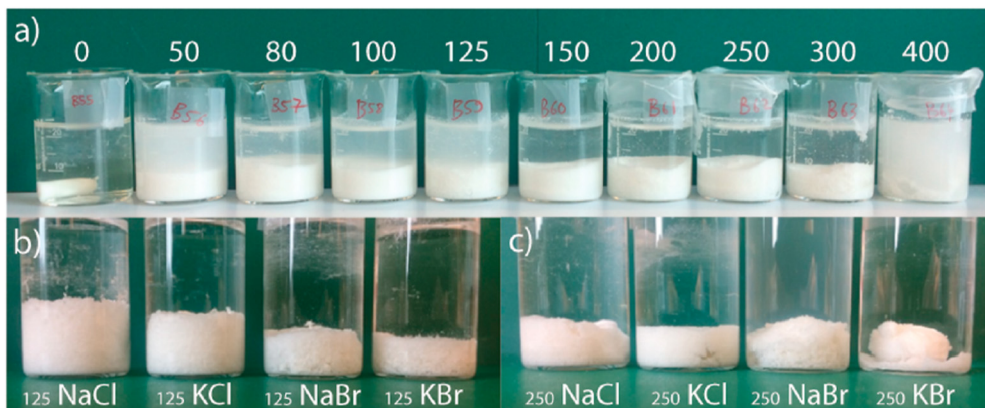


Fig. 9 (a) Effect of NaCl concentration (0–400 mM) on PSS–PDADMAC PEC formation. (b) and (c) Effect of salt type and salt concentration [(b) 125 mM and (c) 250 mM] on PSS–PDADMAC PEC formation – from Krishna *et al.*²¹¹

opportunities associated with the development and application of saloplastic materials.

2.4.1 Energy conversion and storage technologies. Ion exchange membranes (IEMs) and separators play a crucial role in the performance and efficiency of energy conversion and storage devices, such as fuel cells and batteries. These thin, selective barriers are designed to regulate the flow of ions within electrochemical cells, thereby preventing short circuits and ensuring the efficient operation of the devices by minimising the occurrence of unwanted side reactions.^{217–220} Their performance is generally evaluated based on their ionic conductivity and selectivity, which are directly related to the overall efficiency of the electrochemical process.^{217,218} Ionic conductivity refers to a membrane's ability to facilitate ion transport, whereas selectivity describes its capability to selectively allow specific ions to pass while obstructing others. Furthermore, due to the harsh operating conditions typically encountered in electrochemical systems, it is imperative that these membranes exhibit robust chemical, thermal, and mechanical stability.²²¹ Additionally, it is essential to consider sustainability in the

development of new materials, as a significant portion of commercially available membranes contain fluorinated functional groups, which may face future bans, and many solvents used in their production are toxic and not environmentally friendly.^{222,223}

2.4.1.1 Hot-pressed saloplastic ion exchange membranes. To address these requirements, a hot-pressing method has been developed to produce thin, dense, and transparent films from salt-plasticised polyelectrolyte complexes.²¹¹ The hot-pressing method involves placing the solution precipitated PEC in a hot-pressing setup, where it is heated to a predetermined temperature and subjected to pressure for a specified time to achieve plasticisation. After cooling under pressure, the saloplastic film can be removed from the mould. A schematic of this process is given in Fig. 10. When pressing non-stoichiometric PECs, either the polyanion or polycation is present in excess, imparting a negative or positive charge to the film. Consequently, these saloplastic films can function as IEMs. This membrane fabrication method is unique in that it relies solely on salt

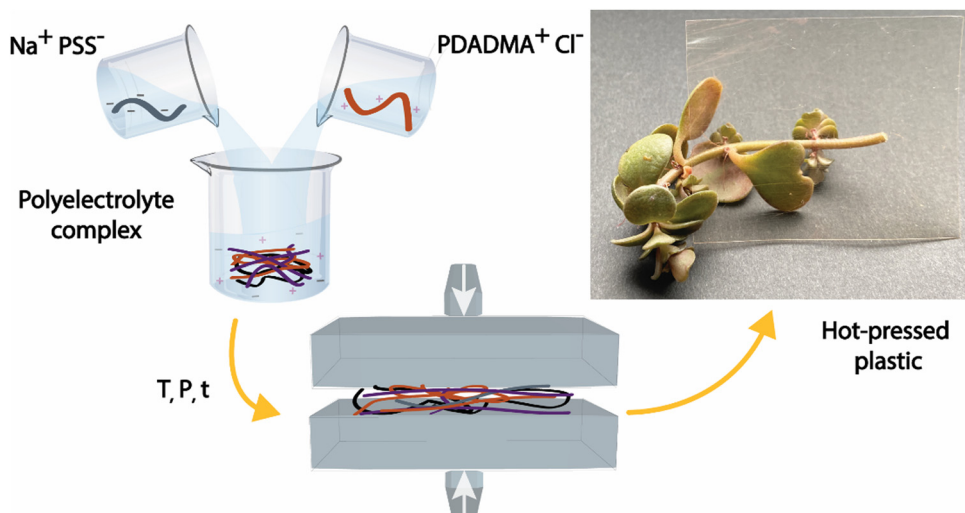


Fig. 10 Schematic of hot-pressing polyelectrolyte complexes into dense saloplastic membranes – from Krishna *et al.*²¹¹



water as a solvent,²²⁴ eliminating the need for harmful chemicals. The membranes are also recyclable,^{10–12} making this a truly sustainable IEM.

A subsequent study demonstrated that the choice of polyelectrolyte pair as well as their stoichiometric ratios could affect the charge density and overall performance of the saloplastic IEMs.²²⁵ It was established that both anion and cation exchange membranes can be produced from the same polyelectrolyte pair by varying the molar ratio during complexation. Given the known chemical stability of PSS-PDADMAC PECs, this system was further investigated to assess its long-term pH stability by comparing the permselectivity of the PSS-PDADMAC anion exchange membrane before and after exposure to extreme pH environments (pH 1 and 14). The results indicate no significant change in permselectivity, confirming the stability of materials across a broad pH range (1–14), suggesting that saloplastic PSS-PDADMAC IEMs may be suitable for fuel cell applications.^{224,225} In a separate study, cation exchange membranes prepared from PSS-PVAm were investigated. Previously, PVA and PVAm were found to be monovalent-selective for K⁺ over Na⁺, which opens up the possibility of removing excess sodium from agricultural feed streams, such as greenhouses, to maintain a healthy balance of salt ions.²²⁶

While significant progress has been made in the development of sustainable, chemically stable saloplastic IEMs, further optimisation is required to improve the charge density of these membranes. This is necessary to improve their performance at higher concentrations, which is currently inferior to that of commercial membranes.²²⁴ This enhancement is critical for improving their performance at higher concentrations, which currently lags behind that of commercial membranes. Additionally, the processing of these membranes largely relies on trial and error to identify optimal hot-pressing conditions (temperature, pressure, and time) for each PEC; in our opinion, efforts are needed to transform PEC processing into a more systematic scientific approach.

2.4.1.2 Thermally compressed PECs as battery separators. A similar approach was used to produce thermally compressed PSS-PDADMAC films to be used as separators for rechargeable zinc–air batteries.²²⁷ In this investigation, PSS-PDADMAC PECs were prepared at various stoichiometric ratios (PDADMAC : PSS = 2:1, 1.5:1, 1:1, 1:1.15, 1:1.5, and 1:2) and then thermally compressed into thin films. By adjusting the molar ratio during complexation, the study successfully produced separators with both negative (excess PSS) and positive (excess PDADMAC) charges, which were qualitatively assessed through selective absorption of anionic and cationic dyes. The performance of these separators was evaluated based on ionic conductivity, selectivity, and performance in rechargeable zinc–air battery applications.

In this application, the separator's role is crucial as it allows the selective permeation of hydroxide ions while preventing crossover of zincate ions, thereby enhancing the battery's efficiency and longevity. The separator with the highest PDADMAC content (2:1) exhibited superior performance by achieving the lowest zincate permeability and the highest ionic conductivity.

Notably, the performance of this separator was comparable to that of a commercially available zinc–air battery separator. Moreover, when conducting galvanostatic cycling testing using a homemade rechargeable zinc–air battery, this separator showed good stability after 150 cycles of testing.²²⁷

This study underscores the significant influence of PEC composition on the performance of the separator. However, the precise charge or exact composition of the PEC remains undetermined, complicating reproducibility. The results also indicate the complexity of achieving the desired stoichiometry, as the 1:1 ratio resulted in a positively charged membrane. Therefore, improved control over charge compensation and enhanced methodologies for quantifying the charge are essential for future research.

2.4.1.3 Functionalised saloplastic films. Li *et al.* investigated an innovative method to enhance the antimicrobial properties of saloplastic films through lysozyme functionalization. By integrating lysozyme into hot-pressed PSS-PDADMAC saloplastic films, the authors aimed to expand the applicability of these materials in sustainable antibacterial applications, such as food packaging, biomedical devices, and microbiological environments. The functionalization process involved a post-treatment procedure that introduced lysozyme into the films through the formation of micropores. These pores were created by annealing the saloplastics in a solution with a high salt concentration, which breaks the ionic crosslinks within the material. The films were subsequently immersed in water, leading to a significant reduction in salinity, which induced pore formation. These pores served as sites for lysozyme entrapment, after which the films were immersed in a salt solution to close the pores. The resulting films demonstrated strong enzymatic activity against *Micrococcus lysodeikticus* and retained 72% of their activity after seven days. This work proves that it is possible to impart biological functionality to saloplastic films, thereby enhancing their potential as sustainable antimicrobial materials.¹⁰³

Saloplastic materials also hold promise in energy storage applications, particularly as electrode materials for supercapacitors. Supercapacitors, known for their rapid charge–discharge rates, rely on the development of thin, flexible films that can effectively store and release energy.²²⁸ Polyelectrolyte complex composite (CPEC) membranes, formed by combining polyelectrolyte complexes (matrix materials) with electroactive fillers, offer a versatile platform for supercapacitor electrodes. While a matrix is not always essential for supercapacitor electrodes, its inclusion can provide structural support and improve the mechanical stability and flexibility of the electrode material. Luangaramvej and Dubas (2021) demonstrated the incorporation of polyaniline, a conductive filler, into a stoichiometrically prepared PSS-PDADMAC polyelectrolyte complex matrix. The resulting salt-plasticized CPECs were formed into membranes *via* compression molding. The inclusion of the PEC matrix allowed the membrane to withstand deformation, whereas pure polyaniline would have been prone to cracking. Furthermore, the CPECs were optimized through *in situ* polymerization of polyaniline, resulting in a membrane that exhibited stable specific capacitance over 2000 charge–discharge cycles. This approach underscores the potential of polyaniline-loaded CPECs



as environmentally friendly materials for supercapacitor electrodes, offering a promising avenue for the development of sustainable energy storage solutions.²²⁸

This section discussed two innovative approaches to functionalizing saloplastic films, showcasing their potential in both antimicrobial and energy storage applications. Together, these studies highlight the versatility of saloplastic materials and their potential for further development and use in specialized, sustainable applications across different industries.

2.4.2 Challenges and opportunities. While several promising applications for saloplastic materials have been investigated, their use on a larger scale is hindered by reproducibility and scalability challenges. Below, we will discuss the main challenges associated with processing saloplastic PECs and their limitations in specific applications and environments due to stability concerns.

2.4.2.1 Challenges in PEC production and processing. (a) Complexation control: reproducibility and scalability of the complexation step is one of the biggest issues facing saloplastic production. Studies have shown that stoichiometric ratios of polyelectrolytes do not consistently yield stoichiometric PECs, with non-stoichiometric complexes often forming despite the intended molar ratios.^{201,225,227} This issue arises because complexation is a kinetically limited process that is influenced by various factors, including polyelectrolyte molecular weight, polyelectrolyte ratio, polyelectrolyte concentration, pH, salt type, ionic strength, solution temperature, mixing order, addition rate, and mixing time.²¹⁵ Scale-up is therefore challenging as it is much more difficult to control complexation when large quantities are produced.¹⁰

(b) Polyelectrolyte variability: variability in polyelectrolytes is a significant factor affecting the reproducibility of complexation. For instance, we have observed variations in solid content (wt%) in PDADMAC across different batches and suppliers. If this variability is not accounted for, the molar ratio at which polyelectrolytes are mixed may be inaccurate, impacting PEC stoichiometry. PDADMAC is also commercially supplied with a broad molecular weight distribution (MWD) with varying PDIs.²²⁹ This could potentially lead to issues with reproducibility since it is known that molecular weight influences complexation.

(c) Processing: PEC processing for saloplastics has traditionally been a trial-and-error process. Research has shown that processing conditions, such as temperature and pressure, are PEC-specific^{211,225} and that the science behind these parameters is not well understood. In our experience, different polyelectrolyte pairs and molar ratios require distinct processing conditions for optimal performance, which can be a time-consuming and challenging process. Furthermore, as previously outlined, the composition of PECs is difficult to control due to the large number of variables influencing complexation. This variability makes them challenging to process into saloplastics, as each batch may have a different composition and characteristics.

2.4.2.2 Stability challenges in different environments and applications. (a) Hydration: the hydration state of saloplastic

materials significantly impacts their mechanical properties. In a dry state, these materials tend to become brittle,¹⁰ which limits their use in applications where they are not maintained in a hydrated environment. If these materials are used in applications where they are not fully immersed in a solution, fluctuations in humidity may impact their mechanical stability. This phenomenon is attributed to the plasticizing effect of water on polyelectrolyte materials. We noticed that the comparatively high water uptake of saloplastic materials negatively affects their performance as ion exchange membranes and separators.²²⁴ High water uptake essentially dilutes the concentration of charges within ion exchange membranes which will allow more undesired ions to pass, thus reducing their selectivity.²³⁰

(b) Salt concentration: saloplastics can also be affected by salt type and concentration. This is because salt dopes PECs, affecting their stability by disrupting ion pairing within the complexes. Consequently, the use of saloplastic ion exchange membranes is currently restricted to processes involving lower salt concentrations to maintain good selectivities.²²⁴ Sudden fluctuations in salt concentration can also result in the formation of micropores^{201,231,232} within the PEC structure, which could impact its structural integrity and functionality. For ion exchange membranes, it is important to have dense films since pores would negatively impact the membranes' selectivity and conductivity.

(c) Temperature: temperature sensitivity is another limitation of saloplastics. The glass transition temperatures of saloplastic materials are highly dependent on factors such as salt concentration and PEC stoichiometry.²¹⁰ Studies have shown that the glass transition of PSS-PDADMAC PECs is relatively close to room temperature,²³³ which severely affects their mechanical stability and limits their use to low-temperature applications.

(d) pH: for strongly charged polyelectrolytes, pH does not affect the charge of the materials. However, exposure to extreme pH environments could affect their stability. Literature has reported that quaternary ammonium groups, which are present in PDADMAC, might be degraded at high pH through Hoffman elimination,²³⁴ nucleophilic substitution²³⁵ or ylide formation.²³⁶ However, studies have shown that PSS-PDADMAC saloplastic membranes exposed to extreme alkaline and acidic environments (pH 1–14) for 30 days did not show signs of degradation.²²⁴

2.4.2.3 Opportunities. To address the challenges associated with PEC production and processing, a deeper understanding of the variables influencing complexation is needed. This knowledge could enable the development of more controlled and reproducible methods for producing PECs at a large scale. Research by Shamoun and colleagues has made important strides in this area, by first developing a ¹H-NMR method to accurately quantify the composition of PSS-PDADMAC PECs and then identifying conditions under which large quantities of stoichiometric PECs can be consistently produced.¹⁰ Their work demonstrated that dilute polyelectrolyte solutions (125 mM) and low salt concentrations (250 mM KBr) lead to the formation of stoichiometric complexes when equimolar solutions of PSS and PDADMAC are added simultaneously in a separate beaker



under stirring.¹⁰ These findings provide a promising starting point for further research aimed at scaling up PEC production and optimizing the complexation process.

Addressing variability in polyelectrolyte materials presents additional challenges, particularly when using commercially available polyelectrolytes. In our opinion, simple quality control measures, such as verifying the solid content of batches, could significantly reduce variability. Furthermore, molecular weight distribution in commercial polyelectrolytes may be improved by fractionation techniques. For instance, one study successfully narrowed the molecular weight distribution of PDADMAC from 3.3 to 1.4 through acetone precipitation and centrifugation, recovering higher molecular weight chains.²²⁹ We think alternative fractionation methods, such as dialysis, could also be employed to reduce the polydispersity of polyelectrolyte solutions.

Further opportunities lie in optimizing the processing conditions of PECs. Previous studies have explored the impact of time, temperature, and salt concentration on PEC behaviour. For example, Shamoun *et al.* observed that PECs doped with higher salt concentrations could be extruded at lower temperatures. This observation led to further investigation into the relationships between processing time, salt doping, and temperature on the thermal behaviour of extruded PSS-PDADMAC PECs. Using dynamic mechanical thermal analysis, the authors measured the T_g of the PECs, noting significant transitions in modulus as the material shifted from a glassy to a rubbery state. The study revealed that glass transition temperature varied with the deformation rate, in accordance with the time/temperature superposition principles, and that higher salt concentrations led to a reduction in glass transition temperature, indicating a plasticizing effect due to the breakage of ion pairs within the complexes. These findings were used to derive an empirical equation that can be used to predict the thermal behaviour of PECs under varying conditions, thus offering valuable insights into how time, salt concentration, and temperature can be manipulated to optimize PEC processing.²³³ Such insights provide valuable guidelines for optimizing PEC processing, offering a shift from trial-and-error experimentation to more systematic, science-driven approaches.

Finally, the application of saloplastic materials is often limited by their sensitivity to environmental factors such as salt concentration, temperature, pH, and hydration. While some of these challenges may be difficult to fully overcome, modifying saloplastic materials through crosslinking presents a promising strategy to enhance their performance. Crosslinking introduces covalent bonds between polymer chains, which can reduce chain mobility, thereby improving the material's mechanical stability, thermal resistance, and reduced swelling.^{34,237–239}

To summarise, the development of saloplastic materials offers significant potential but faces challenges related to reproducibility, scalability, and environmental stability. However, opportunities to address these challenges exist, driven by an improved understanding of material properties, processing conditions, and material modifications. By adopting a more systematic, science-driven approach, a lot of these challenges can be overcome.

2.5 PECs as aqueous two-phase systems

Polyelectrolyte complex phases and their supernatants are aqueous two-phase systems. Indicated in Scheme 1 are the complex coacervate and its supernatant and the PEC and its dilute phase. Since molecules can distribute among these phases, these systems can potentially be used to extract, separate and/or purify molecules.^{240–242} To effectively study the uptake of molecules in aqueous two-phase systems based on polyelectrolyte complex phases, develop new applications, or improve materials derived from the PEC phase, a clear understanding of both the dense and dilute phases is essential. Particular attention should be given to the partitioning behaviour of molecules within these systems, as this plays a crucial role in their performance. Protein partitioning, in particular, is a key factor that significantly influences the properties and functionalities of PECs, as proteins may interact differently across phases depending on their charge, structure, and environmental conditions.

The partitioning of molecules plays a pivotal role in cellular life, enabling the organization and regulation of biochemical processes in response to changing conditions. Membrane-less organelles (MLOs) are prime examples of this mechanism in action, acting as dynamic, spatiotemporal hubs that coordinate and control cellular activities. Unlike traditional organelles bound by membranes, MLOs form through liquid–liquid phase separation, allowing them to compartmentalise specific biomolecules and create distinct microenvironments within the cytoplasm. The ability to assemble and disassemble in response to cellular signals enables MLOs to regulate the biochemical reactions on demand, ensuring both precise timing and spatial organization of processes critical for maintaining homeostasis and adapting to environmental changes.^{243,244}

The unique behaviour of MLOs, their capacity to phase-separate, partition specific molecules, and dynamically reorganize, presents an exciting model for bio-inspired materials design. One promising approach for mimicking these natural systems relies on polyelectrolyte complexation. These complexes exhibit the potential to create compartmentalized environments, making them useful in applications that require the selective partitioning of molecules. By leveraging the principles underlying MLOs, polyelectrolyte complexes could be designed to achieve selective extraction, separation, and concentration of target molecules in non-biological settings.

Water and wastewater treatment is a particularly promising field for applying polyelectrolyte complexes. While polyelectrolytes have been used as additives in coagulation and flocculation processes to assist in potable water and sludge dewatering, PECs offer a more direct and potentially more efficient alternative. In fact, PECs can extract contaminants from water, bypassing the need for inorganic coagulants commonly used in traditional processes.²⁴⁵

To improve and develop novel applications, a deeper understanding of the underlying mechanisms governing polyelectrolyte complexation is crucial. This might include a thorough characterization of polyelectrolyte systems, such as understanding the intrinsic properties of the individual polyelectrolytes *i.e.* response to environmental factors such as pH and ionic



strength. These properties influence how polyelectrolytes interact with each other and with other molecules. Furthermore, a detailed investigation of the complexation process is needed, such as the kinetics of complex formation, the stability of the complexes, and their ability to encapsulate and release specific molecules. Moreover, the collection of extensive data sets on these interactions will enable the development of predictive models. Such models can optimize the design of polyelectrolyte complexes for specific applications, such as targeting specific molecules or achieving desired levels of extraction efficiency. These insights can lead to the development of systems with improved performance, scalability, and environmental compatibility. We believe that by integrating the principles of MLOs' behaviour into PECs, we open the door to innovative materials that can mimic the ability of nature to organize and partition molecules with precision.

Building on this foundation, the following section will delve into the uptake and release of molecules, specifically proteins, in polyelectrolyte complexes and complex coacervates. By examining successful systems and identifying key parameters for optimization, we aim to advance the use of PECs as aqueous two-phases for separation.

2.5.1 Protein encapsulation and extraction. Proteins are highly complex macromolecules consisting of chains of amino acid residues, that play a critical role in biological processes, spacing from catalysis, structural function, transport, and storage, to signal transduction. Chemically, proteins are ampholytic compounds, containing both acidic (carboxyl) and basic (amino) moieties within their amino acid residues. Depending on the pH of their environment, these groups can carry a positive or negative charge. This amphoteric nature facilitates interactions with various molecules and ions, underpinning their different functions within living organisms.²⁴⁶

Proteins exhibit a high degree of structural organization, characterised by four levels: primary, which refers to the linear sequence of amino acids; secondary, which involves the local folding into α -helices and β -sheets; tertiary, which describes the three-dimensional shape of a single polypeptide chain; and quaternary, which pertains the assembly of multiple polypeptide subunits into a functional complex.²⁴⁷

Due to their sensitivity, proteins are prone to denaturation, a process where they lose their functional structure when exposed to changes in pH, temperature, or chemicals. This instability outside their natural environment presents significant challenges for storage and practical applications. Therefore, developing effective strategies for both extraction and encapsulation that maintain protein stability is essential for advancing protein-based uses.^{248,249} Both polyelectrolytes and complex coacervates can be used to stabilise proteins in aqueous environments, avoiding the use of solvents that can potentially destabilise these molecules.

As Scheme 1 illustrates, the phenomenon of polyelectrolyte complexation is influenced by several factors, including polyelectrolyte architecture, pH, salt concentration, ionic strength, mixing order and speed, and temperature. These factors influence the final morphology and composition of PECs, allowing

for the formation of structures ranging from liquid-like to solid-like complexes.

Proteins, as a special type of polyelectrolyte, exhibit charge behaviour that is dependent on their isoelectric point (pI), *i.e.* the pH at which the net charge is zero (refer to Fig. 1b). At pH values above the pI, proteins carry a net-negative charge, whereas below the pI, they bear a net-positive charge. Since proteins can be considered polyelectrolytes, they can form polyelectrolyte complexes with oppositely charged (synthetic) polyelectrolytes. Here we focus on three component systems where two oppositely charged polyelectrolytes form the PEC or complex coacervate and proteins are taken up by this complex phase. This variability in the charge of the protein as a function of pH is crucial when designing protein extraction or encapsulation methods using polyelectrolyte complexation, as it directly impacts loading and release efficiency.

By controlling factors such as pH, salt concentration, and ionic strength, the extraction and release of molecules^{250,251} from PECs can be finely tuned. Specifically, increasing salt concentration disrupts intrinsic ion pairs, converting them into extrinsic ion pairs, which loosens the PEC matrix and promotes the release of extracted molecules.^{252–254} Adjusting the pH, on the other hand, modifies the charge density of the polyelectrolytes themselves (for weak polyelectrolytes), either enhancing or reducing complexation strength and can thus trigger molecule release at specific pH thresholds (see Fig. 1b). Together, these parameters allow for targeted control over PEC composition, regulating both the encapsulation and release of molecules from the complex. If the encapsulation process is selective towards one specific species in the solution, PECs could be used for extraction, and the controlled release could then be called back-extraction.

Several polyelectrolyte complex systems have been used to encapsulate proteins.^{255–260} Perry *et al.* investigated the encapsulation of a range of proteins with different charges within a poly(L-lysine) and poly(D,L-glutamate) system across different solution conditions and polymer properties. They found that proteins with clustered-like charged residues showed enhanced uptake, resulting in increased sensitivity of the system to solution conditions.²⁵⁵ In our group, we found that the concentration of the lysozyme inside the complex can become so high that proteins irreversibly aggregate inside the complex.²⁵⁸ In addition, we observed that the uptake of lysozyme and a chemically modified version of this protein strongly depends on the stoichiometry of the complex phase.^{258–260} This selectivity in protein uptake can be used to separate proteins from protein mixtures. A study by Van Lente *et al.* explored the extraction of lysozyme from a multi-protein mixture using four PEC systems,²⁶⁰ each composed of different weak and strong polyelectrolyte combinations. Fig. 11 shows the partitioning profiles of lysozyme across these PEC systems at pH 7. While all systems exhibited similar lysozyme uptake profiles as a function of the complex compositions, significant differences emerged during the release phase.

The process of releasing proteins into a new supernatant is referred to as back-extraction. Lysozyme release was induced either by salt addition or pH reduction (Fig. 12). Adding salt



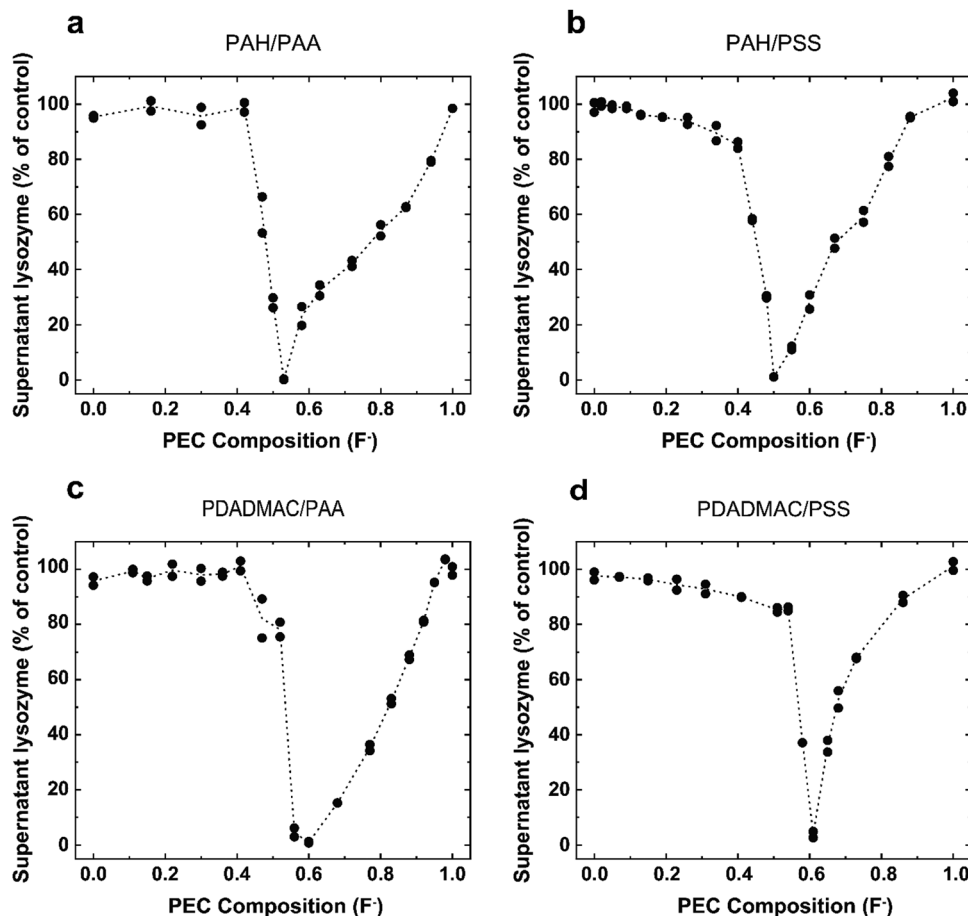


Fig. 11 The partitioning profiles of lysozyme in various PECs at pH 7. (a) Weak/weak PAH/PAA, (b) weak/strong PAH/PSS, (c) strong/weak PDADMAC/PAA, and (d) strong/strong PDADMAC/PSS, at different PEC compositions as expressed in F^- . A low supernatant lysozyme content corresponds to a high PEC lysozyme content and vice versa. Values represent individual measurements; lines connect averages of duplicates. Adapted with permission.²⁶⁰ Copyright 2022, Wiley.

increases charge screening within PECs, weakening the intrinsic electrostatic interactions, loosening the PEC structure, and promoting protein release (Fig. 12a). In contrast, Fig. 12b shows lysozyme release through pH reduction, which alters the charge properties of the polyelectrolytes. Systems composed of strong polyelectrolytes, like PDADMAC/PSS, remain more stable under varying conditions, whereas PECs containing weak polyelectrolytes or combinations of weak and strong polyelectrolytes disintegrate more easily under pH changes. This effect occurs due to changes in the degree of ionization of charged groups on both the polyelectrolytes and proteins, which affect the overall stability of the PEC structure. Additionally, salt and pH variations cause PECs to swell or shrink, sometimes resulting in structural transformations such as hydrogel formation in PDADMAC/PAA PECs at low pH.²⁶⁰

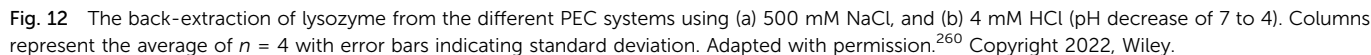
These findings emphasize how the properties of the polyelectrolyte complex, along with environmental conditions, can significantly affect the efficiency of protein extraction. Moreover, the back-extracted lysozyme retained its enzymatic activity, proving the capability of PEC systems to function as extraction media for proteins. These findings underscore the broad applicability of PECs for protein extraction, particularly in fields such

as biotechnology and wastewater treatment, where selective separation is essential.

Overall, polyelectrolyte complexation is a versatile and tuneable method applicable to both extraction and encapsulation of proteins. While extraction through PEC focuses on the selective separation of proteins from complex mixtures based on electrostatic interactions, encapsulation offers a way to stabilize and protect proteins for further use. Both processes are essential for advancing applications in biotechnology, pharmaceuticals, and environmental science.

In addition to the extraction and encapsulation processes, the reversibility and long-term stability of PECs are critical considerations, especially for applications requiring multiple uses. PEC systems that can undergo reversible complexation, by adjusting factors like pH or ionic strength, allow for the recovery of both the extracted protein and the polyelectrolytes. Studies, such as those by Van Lente *et al.*, have demonstrated the controlled release of proteins and other molecules by altering the composition and environmental conditions, suggesting that PEC could be recycled for multiple rounds of extraction.^{258,259,261} However, understanding how to optimize the balance between stability and reversibility is an important





In Scheme 1 we summarised the different controlling parameters per step for the production of PEC-based materials. Depending on the end material different challenges have been identified per step. For all the materials the properties of the starting polyelectrolytes should be known (step 1, Scheme 1). For the preparation of porous membranes, the viscosity of the casting solution is important for the kinetics of the phase separation and eventually affects the porosity of the membrane. Especially when biopolyelectrolytes are used, batch-to-batch variation could have an impact on the reproducibility of the

membrane production process.⁹⁵ The starting polyelectrolytes are either in powder form or can be purchased as solutions. In the powder form, one has full control of the concentration of the polyelectrolyte solution. When the polyelectrolytes are delivered as solutions, it is important to check whether the concentration on the label is correct. In our experience, if the label says 20 wt% this can vary between 18–22 wt%. Since the mixing ratio of the polyelectrolytes is important for the characteristics of the final material, assuming that the concentration of the polyelectrolytes is correct could result in reproducibility issues. In addition, the polydispersity of the polyelectrolytes could affect the final PEC-material, and methods to fractionate polyelectrolytes have been proposed.²²⁹

For step 2 in Scheme 1, it is important to record all the parameters in sufficient detail. If weak polyelectrolytes are part of the system, measuring the pH before and after complexation (in step 3) is advisable since charge regulation is expected to occur in these systems.^{19–21} Ion-specific interactions between the polyelectrolytes could play a role in this step. The temperature as a parameter during polyelectrolyte complexation has not been extensively explored, but it is known that it may well affect the properties of the polyelectrolyte complex²²⁵ and the properties of the material after processing.¹⁰³

The final composition of the PEC in terms of complex charge-stoichiometry is strongly affected by the mixing conditions. Some polyelectrolyte pairs might even have a preferential non-stoichiometric composition.^{201,224,225,227,264} To avoid human errors and enhance reproducibility flow cells could be possibly used to produce PECs. This approach could be coupled with machine learning algorithms to optimize PEC composition in real-time based on immediate feedback from characterization data. To understand how the way of mixing influences the final PEC a full characterisation of the system is required. Below we give an overview of the different characterization methods that have been used to characterise PECs and the final PEC-based materials.

3.1 Characterization techniques

To fully understand the mechanisms behind polyelectrolyte complexation and the partitioning of molecules, it is crucial to use effective characterization techniques. These methods not only allow for the detection and quantification of the overall composition of PECs in terms of polyelectrolytes and counterions but also provide valuable insights into the mechanism of the uptake of proteins and other molecules under different environmental conditions. The following section will outline the most commonly used techniques for characterizing both the overall composition of PECs and the protein content, with a particular focus on how these methods can be used to generate detailed data sets that can be used for multiscale modelling, machine learning and AI.

3.1.1 Complex composition, polyelectrolyte, and counterion quantification. Understanding the full composition of polyelectrolyte complexes is crucial for optimizing their performance. Advanced techniques, including attenuated total reflection–Fourier transform infrared spectroscopy (ATR), thermal

gravimetric analysis (TGA), fluorescence spectroscopy, and nuclear magnetic resonance (NMR), provide detailed insights into polyelectrolyte and counterion quantification.

3.1.1.1 Attenuated total reflection–Fourier transform infrared spectroscopy. Attenuated total reflection–Fourier transform infrared spectroscopy (ATR–FTIR) is an effective technique for characterizing polyelectrolyte complexes. This method allows for the assessment of the degree of ionization in polyelectrolytes or protonation levels^{265,266} in the case of weak polyelectrolytes, which is a key factor in complex formation. Typically, studies utilize ATR–FTIR to investigate how the degree of ionization varies with the pH of the solution.^{200,266–268}

While ATR–FTIR is a valuable tool for characterizing polyelectrolyte complexes, it has some limitations. Compared to other techniques, ATR–FTIR offers more qualitative rather than quantitative analysis. Additionally, the technique is sensitive to experimental conditions such as temperature, humidity, and pressure, which can influence the results, given that polyelectrolytes are often affected by environmental factors. Furthermore, the presence of functional groups with similar vibrational frequencies can lead to overlapping peaks, complicating the process of peak assignments. In summary, ATR–FTIR is a useful analytical tool for polyelectrolyte complex characterisation, especially to obtain information about the degree of ionisation in the PEC, but it should be used in conjunction with other techniques to provide a more comprehensive and quantitative understanding of these systems.²⁶⁹

3.1.1.2 Thermal gravimetric analysis. Thermal gravimetric analysis (TGA) is another method for quantifying both polymers and salt content in polyelectrolyte complexes.³⁰ TGA measures changes in the physical and chemical properties of a sample as a function of increasing temperature or time, making it applicable for analysing complex phases. However, TGA has several limitations. It primarily provides data on mass loss related to degradation and thermal stability but does not offer insights into molecular interactions or the degree of complexation. The thermal decomposition of the components in the complex can overlap, complicating the differentiation of contributions from individual polyelectrolytes, and counterions. Additionally, since polyelectrolyte complexes are often hygroscopic, absorbed water can cause mass loss at low temperatures, which could hide thermal events associated with the decomposition of the compound itself. Moreover, TGA is a destructive technique, meaning samples cannot be preserved post-analysis. Therefore, to achieve a more comprehensive chemical analysis, it is advisable to pair TGA with complementary techniques, such as NMR, which can provide additional specificity.

3.1.1.3 Nuclear magnetic resonance. NMR has proven to be a versatile technique for characterizing polyelectrolytes and their counterions.²⁷⁰ For instance, ¹H-NMR has been used to determine the mole ratio of polyelectrolytes in the dense phase by redissolving dried complexes in high salt solutions when using strong polyelectrolytes²⁶⁴ or in a mixture of an acid or base in a high salt concentration solution if weak polyelectrolytes were



used, to protonate or deprotonate one of the polyelectrolytes.²⁶⁷ Currently, ¹H-NMR allows for the quantification of individual polyelectrolytes and their mole ratios in both the dense and supernatant phases, with the latter phase being diluted with D₂O.^{271–277}

To gain a comprehensive understanding of the entire system, it is essential to quantify the inorganic content alongside the organic one. Compared to more sophisticated techniques such as radiolabelling^{38,231} and neutron activation analysis,²⁷⁸ which involve extensive preparation steps and longer processing times, NMR offers a non-invasive, *in situ*, and rapid approach. It allows the analysis of various NMR-active nuclei,^{277,279} including Na and Cl, through ²³Na and ³⁵Cl-NMR, respectively.^{280,281} This makes it possible to quantify the inorganic content in polyelectrolyte systems.

Since polyelectrolyte complexation is entropically driven, the majority of counterions are expected to be in the supernatant phase, with a minor amount remaining in the dense phase. Accurate quantification of counterions provides valuable insight into the mechanisms underlying protein partitioning within polyelectrolyte complexes.²⁸²

NMR does not face the same challenges related to the physical state of the sample as other techniques. Nevertheless, it is recommended to conduct experiments at high polyelectrolyte concentrations to enhance the formation of a denser phase and facilitate phase separation. An advanced approach would involve performing complexation directly within the NMR tube, which could minimize errors associated with sample separation.

To conclude, NMR, along with the previously discussed techniques, offers valuable insights into the composition and behaviour of PECs. However, each method has its limitations, and till now, no single technique provides a complete picture. Therefore, the most comprehensive understanding of PEC systems comes from using a combination of these methods, allowing researchers to explore protein partitioning, polyelectrolyte interactions, and counterion behaviour with greater precision. As the field advances, further refinement of these characterization techniques will be crucial for optimizing PEC-based applications in protein extraction, encapsulation, and other emerging technologies.

3.1.2 Protein quantification. To use PECs as extraction media, in addition to the quantification of polyelectrolytes and counterions, the quantification of the protein content is also key. Understanding the balance between these components is essential for gaining insights into the structure and the stability of PECs, thereby enabling the development of systems tailored for specific applications.

To accurately characterize the extraction behaviour of polyelectrolyte complexes, methods are required to determine which molecules are present in each phase. For protein partitioning specifically, UV-vis spectroscopy is a common approach, as it detects the characteristic absorbance peak of tyrosine and tryptophan residues at 280 nm.^{258–260,283} However, while effective for high protein concentrations, this method is less sensitive at lower concentrations. Additionally, if polyelectrolytes contain aromatic or conjugated groups, their UV absorbance may interfere, complicating accurate protein quantification.²⁸⁴

Colorimetric assays offer another option for protein quantification, particularly useful when proteins lack distinctive signals or are present at low concentrations.²⁸⁵ These assays rely on the interaction between a dye and specific amino acid residues within the protein, though they cannot differentiate between multiple proteins within a sample.^{255,286}

For more sensitive detection, especially at low protein concentrations, fluorescence spectroscopy is effective when proteins are intrinsically fluorescent or labelled with fluorescent tags, such as GFP.^{287–289} This approach can also apply to other molecules. For instance, Spruijt *et al.* used a fluorescent label on one polyelectrolyte to measure its distribution between the polymer-rich phase and the supernatant.²⁹⁰ However, introducing fluorescent labels can potentially influence the complexation process itself, and fluorescence measurements have additional limitations. Fluorescence is highly dependent on the local environment, which may vary significantly within PECs. Therefore, to accurately determine the composition of the phases without affecting complexation, it is preferable to use less invasive methods.

Typically, protein quantification in polyelectrolyte complexes is performed by separating the dense and the supernatant phases, either through pipetting or decantation, followed by individual analysis. The partition coefficient is then calculated as the ratio of the concentration of protein in the dense phase divided by the concentration of protein in the supernatant. Despite the straightforward nature of these techniques, they present certain challenges. For solid-like complexes, the dense phase is hard to characterize using these methods due to its physical state.

In the case of liquid-like complexes, the small volume of the coacervate, often in the range of a few microliters, complicates the quantifications and necessitates assumptions or approximations. One approach to address this issue relies on calculating the mass balance of the system, where the protein mass in the coacervate phase is inferred by subtracting the protein mass in the supernatant from the total protein mass added.²⁵⁵ To overcome these limitations, it is essential to explore new techniques that account for these physical constraints, thereby improving the accuracy of protein quantification in polyelectrolyte complexes.

3.1.3 Challenges and opportunities. Scheme 1 illustrates the huge parameter space for polyelectrolyte complexation, which is challenging to map comprehensively. Existing characterization techniques, while valuable, often lack the capacity to fully define PEC composition and stability, making complementary methods essential for a more comprehensive understanding.

PEC-based materials offer sustainable alternatives to conventional materials as they can be produced in water and are recyclable. To get these materials to the next level concise experimental open-access datasets are needed that can optimise polyelectrolyte complexation theory and serve as training data for machine learning approaches and AI. These datasets should contain as much information as possible about how the samples are being prepared and characterised. The parameters mentioned in Scheme 1 can be used as a guide (see Fig. 13). An accurate description of as many parameters as possible should



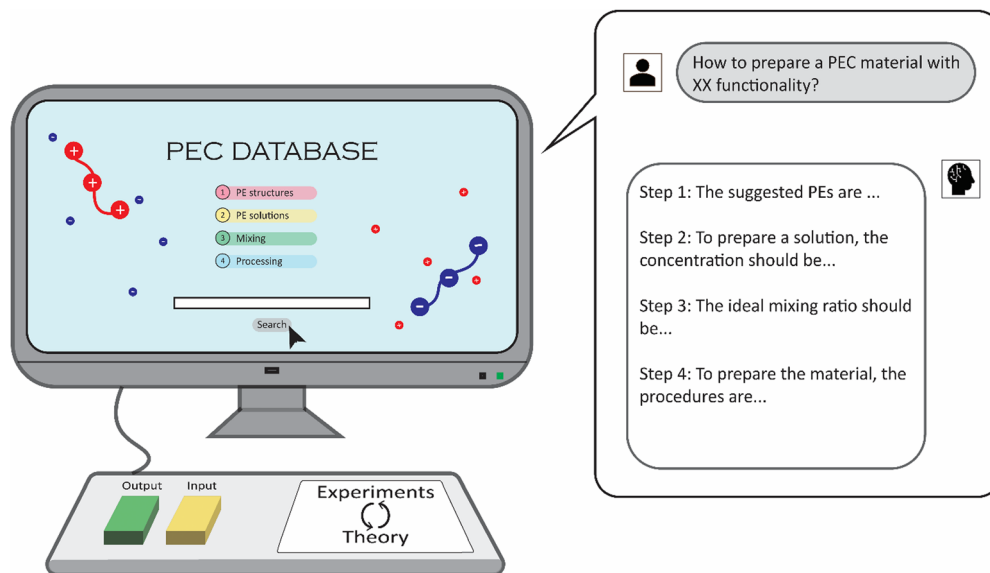


Fig. 13 An example of an AI-assisted PEC database. Depending on the commands to achieve specific PEC materials, all suggested parameters and processing methods could be provided. This database should be consistently updated using both experimental data and theoretical simulations.

be provided in a metafile. Ideally, these data sets should be peer-reviewed by experimentalists and theoreticians and could be shared *via* a trusted database.

4. Summary and outlook

In this review, we discussed how different starting points, as indicated in Scheme 1 as being homogenous solutions, complex coacervates, and polyelectrolyte complexes, can be used to make a variety of materials. Many of these materials were first proposed in 1965 by Michaels. Sixty years later, the field has advanced, but each of the discussed materials, porous membranes, coatings, underwater adhesives, dense ion-exchange membranes, and PEC-based aqueous two-phase systems, presents its own challenges and opportunities.

A key challenge common to all materials is achieving reproducibility. Factors contributing to this challenge include the variability in starting materials, such as incorrect weight percentages or biological variations when biopolyelectrolytes are used. Additionally, differences in sample mixing methods or parameter adjustments can further impact reproducibility. As discussed, numerous factors influence polyelectrolyte complexation, creating a vast parameter space that is difficult to explore experimentally.

In our opinion, polyelectrolyte complex-based materials are particularly promising, as they can be processed in water and have the potential to replace current materials that either contain harmful molecules or are produced through environmentally damaging processes. PEC-based materials, such as porous membranes and aqueous two-phase systems, can be used to purify water, while ion-exchange membranes have potential applications in electrochemical cells. These materials could play a key role in achieving the sustainable development goals 6 (clean water and sanitation) and 7 (affordable and clean energy).

To unlock the full potential of PEC-based materials and advance the field, we envision that recent developments in multiscale modelling, machine learning, and AI could be highly beneficial for their design and prediction. Achieving this will require collaboration between experimentalists and theoreticians to generate detailed experimental datasets, including metadata describing the experimental conditions, varied parameters, and fixed parameters. These datasets can serve as input to validate and improve the theoretical models. To facilitate this, it would be beneficial to establish standardized, peer-reviewed open-access databases for data sharing.

List of abbreviations

7-HC	7-Hydroxycoumarin
A [−]	Negative counterion
AFM	Atomic force microscopy
Ag ⁺	Silver(I) ion
AI	Artificial intelligence
APS	Aqueous phase separation
ATR-FTIR	Attenuated total reflection-Fourier transform infrared spectroscopy
BOPP	Biaxially orientated polypropylene
BSA-FITC	Fluorescein isothiocyanate labelled bovine serum albumin
CaCl ₂	Calcium chloride
C _{salt}	Salt concentration
C _{salt} [*]	Critical salt concentration
CMC	Carboxymethyl cellulose
CMCNa	Carboxymethylcellulose sodium
CPECs	Polyelectrolyte complex composite membranes
CS	Chitosan
CuO	Copper oxide
DMAC	Dimethyl acetamide



DMF	Dimethyl formamide	pK_a	Acid dissociation constant
DMTA	Dynamic mechanical thermal analysis	PLL	Poly(L-lysine)
DOPA	3,4-Dihydroxy-L-phenylalanine	PMADAP	Poly(<i>N,N</i> -[(dimethylamino)propyl]methacrylamide)
DP	Degrees of polymerization	PMMA	Polymethyl methacrylate
ECM	Extracellular matrix	PNIPAM	Poly(<i>N</i> -isopropylacrylamide)
GA	Glutaraldehyde	Pol^+ , Pol^-	Polycation, polyanion
GO	Graphene oxide	PP	Polypropylene
HA	Hyaluronan	PS	Polystyrene
HA	Hyaluronic acid	PSaMA	Polystyrene- <i>alt</i> -maleic acid
IEMs	Ion exchange membranes	PSP	Poly(phosphate sodium) salt
ITO	Indium tin oxide	PSPMA	Poly(3-sulfopropyl methacrylate)
K^+	Potassium ion	PSS	Poly(sodium 4-styrene sulfonate)
KBr	Potassium bromide	PSS	Poly(4-styrenesulfonic) acid
KCl	Potassium chloride	PT	Pectin
LbL	Layer-by-layer	PTFE	Polytetrafluoroethylene
LCST	Lower critical solution temperature	PVAm	Polyvinyl amine
LS	Lignosulfonate	PVA-PAAm	Poly(vinyl alcohol)-polyacrylamide
LSI	Laser speckle imaging	PVP	Polyvinyl pyrrolidone
$L-SO_3Na$	Sodium lignosulfonate	QCS	Chitosan quaternary ammonium salt
M^+	Positive counterion	QP4VP	Polycation poly(<i>N</i> -methyl-4-vinylpyridinium iodide)
MADQUAT	Poly(2-(trimethylamino)ethyl methacrylate)	QVP-C2	Poly(<i>N</i> -ethyl-4-vinylpyridinium)
$MgCl_2$	Magnesium chloride	RH	Relative humidity
$MgSO_4$	Magnesium sulphate	S^+ , S^-	Soluble PECs
MLOs	Membrane-less organelles	SA	Sodium alginate
MW	Molecular weight	SEM	Scanning electron microscopy
MWD	Molecular weight distribution	TA	Tannic acid
Na^+	Sodium ion	Tf_2N	Bis(trifluoromethane)sulfonamide
NaCl	Sodium chloride	T_g	Glass transition temperature
NIPS	Non-solvent induced phase separation	TGA	Thermal gravimetric analysis
NMP	<i>N</i> -Methyl pyrrolidone	TiO_2	Titanium oxide
NMR	Nuclear magnetic resonance	VEGF	Vascular endothelial growth factor
OEGMA	Oligo([ethylene glycol]methyl ether methacrylate)	VOCs	Volatile organic compounds
OTR	Oxygen transmission rate	ϵ -PL	ϵ -Poly-L-lysine
P4VP	Poly(4-vinyl pyridine)		
PAA	Poly(acrylic acid)		
PAE-Cl	Polyamidoamine-epichlorohydrin		
PAH	Poly(allylamine)/poly(allylamine hydrochloride)		
PAMPS	Poly(2-acrylamido-2-methylpropanesulfonic acid)		
PBS	Phosphate-buffered saline		
PC	Polycarbonate		
PDADMAC	Poly(diallyldimethylammonium chloride)		
PDAMAPAA	Poly(dimethylaminopropyl acrylamide)		
PDI	Polydispersity		
PDMAEMA	Poly(2-(dimethylamino)ethyl methacrylate)		
PDMS	Polydimethylsiloxane		
PE	Polyethylene		
PE^+ , PE^-	Polycation, polyanion		
PEDOT	Poly(3,4-ethylenedioxythiophene)		
PEC(s)	Polyelectrolyte complex(es)		
PEI	Polyethyleneimine		
PEM(s)	Polyelectrolyte multilayer(s)		
PET	Poly(ethylene terephthalate)		
p_I	Isoelectric point		

Data availability

The review entitled “Polyelectrolyte complex-based materials for separations, progress, challenges and opportunities” by Jiaying Li, Lijie Li, Hestie A. Brink, Giulia Allegri and Saskia Lindhoud does not contain any new data. Only previously published work has been cited.

Conflicts of interest

The authors declare that there are no conflicts of interest regarding this article.

Acknowledgements

The PhD project of Hestie Brink is funded by NWO. Lijie Li appreciates the China Scholarship Council (CSC) for providing a scholarship.



References

- 1 F. W. Tiebackx, Z. *Chem. Ind. Kolloide*, 1911, **8**, 198–201.
- 2 H. G. Bungenberg de Jong and H. R. Kruyt, *Proc. K. Ned. Akad. Wet.*, 1929, **32**, 849–856.
- 3 C. G. de Kruif, F. Weinbreck and R. de Vries, *Curr. Opin. Colloid Interface Sci.*, 2004, **9**, 340–349.
- 4 T. Moschakis and C. G. Biliaderis, *Curr. Opin. Colloid Interface Sci.*, 2017, **28**, 96–109.
- 5 R. M. Fuoss and H. Sadek, *Science*, 1949, **110**, 552–554.
- 6 J. T. G. Overbeek and M. J. Voorn, *J. Cell. Comp. Physiol.*, 1957, **49**, 7–26.
- 7 A. S. Michaels, *Ind. Eng. Chem.*, 1965, **57**, 32–40.
- 8 P. Schaaf and J. B. Schlenoff, *Adv. Mater.*, 2015, **27**, 2420–2432.
- 9 R. A. Ghostine, R. F. Shamoun and J. B. Schlenoff, *Macromolecules*, 2013, **46**, 4089–4094.
- 10 R. F. Shamoun, A. Reisch and J. B. Schlenoff, *Adv. Funct. Mater.*, 2012, **22**, 1923–1931.
- 11 J. Y. Li, G. van Ewijk, D. J. van Dijken, J. van der Gucht and W. M. de Vos, *ACS Appl. Mater. Interfaces*, 2021, **13**, 21844–21853.
- 12 T. Yuan, X. M. Cui, X. K. Liu, X. X. Qu and J. Q. Sun, *Macromolecules*, 2019, **52**, 3141–3149.
- 13 F. Luo, T. L. Sun, T. Nakajima, T. Kurokawa, A. Bin Ihsan, X. F. Li, H. L. Guo and J. P. Gong, *ACS Macro Lett.*, 2015, **4**, 961–964.
- 14 A. Reisch, E. Roger, T. Phoeung, C. Antheaume, C. Orthlieb, F. Boulmedais, P. Lavalley, J. B. Schlenoff, B. Frisch and P. Schaaf, *Adv. Mater.*, 2014, **26**, 2547–2551.
- 15 J. K. Bediako, E. S. M. Mouele, Y. El Ouardi and E. Repo, *Chem. Eng. J.*, 2023, **462**, 142322.
- 16 S. Manoj Lalwani, C. I. Eneh and J. L. Lutkenhaus, *Phys. Chem. Chem. Phys.*, 2020, **22**, 24157–24177.
- 17 D. Kovacevic, S. van der Burgh, A. de Keizer and M. A. Cohen Stuart, *Langmuir*, 2002, **18**, 5607–5612.
- 18 W. M. de Vos and S. Lindhoud, *Adv. Colloid Interface Sci.*, 2019, **274**, 102040.
- 19 S. Lindhoud, W. Norde and M. A. Cohen Stuart, *J. Phys. Chem. B*, 2009, **113**, 5431–5439.
- 20 V. Pryamitsyn and V. Ganesan, *J. Chem. Phys.*, 2015, **143**, 164904.
- 21 H. Wu, J. M. Ting and M. V. Tirrell, *Macromolecules*, 2020, **53**, 102–111.
- 22 H. H. Hariri, A. M. Leahaf and J. B. Schlenoff, *Macromolecules*, 2012, **45**, 9364–9372.
- 23 M. Tirrell, *ACS Cent. Sci.*, 2018, **4**, 532–533.
- 24 R. Zhang, Y. Zhang, H. S. Antila, J. L. Lutkenhaus and M. Sammakorpi, *J. Phys. Chem. B*, 2017, **121**, 322–333.
- 25 A. D. Kulkarni, Y. H. Vanjari, K. H. Sancheti, H. M. Patel, V. S. Belgamwar, S. J. Surana and C. V. Pardeshi, *Artif. Cells, Nanomed., Biotechnol.*, 2016, **44**, 1615–1625.
- 26 S. Lankalapalli and V. R. Kolapalli, *Indian J. Pharm. Sci.*, 2009, **71**, 481–487.
- 27 V. S. Meka, M. K. G. Sing, M. R. Pichika, S. R. Nali, V. R. M. Kolapalli and P. Kesharwani, *Drug Discovery Today*, 2017, **22**, 1697–1706.
- 28 B. P. Das and M. Tsianou, *Adv. Colloid Interface Sci.*, 2017, **244**, 71–89.
- 29 C. E. Sing and S. L. Perry, *Soft Matter*, 2020, **16**, 2885–2914.
- 30 Q. Wang and J. B. Schlenoff, *Macromolecules*, 2014, **47**, 3108–3116.
- 31 A. E. Neitzel, G. X. De Hoe and M. V. Tirrell, *Curr. Opin. Solid State Mater. Sci.*, 2021, **25**, 100897.
- 32 Y. Liu, M. Zhu, Q. Zhao, Q. An, J. Qian, K. Lee and J. Lai, *J. Membr. Sci.*, 2011, **385–386**, 132–140.
- 33 C.-A. Ghiorghita, F. Bucatariu and E. S. Dragan, *Mater. Sci. Eng., C*, 2019, **105**, 110050.
- 34 S. H. Lee, *Polym. J.*, 2000, **32**, 716–721.
- 35 B. Han, T. Ma, J. H. Vergara, G. R. Palmese, J. Yin, D. Lee and L. Han, *RSC Adv.*, 2017, **7**, 53334–53345.
- 36 E. Diamanti, N. Muzzio, D. Gregurec, J. Irigoyen, M. Pasquale, O. Azzaroni, M. Brinkmann and S. E. Moya, *Colloids Surf., B*, 2016, **145**, 328–337.
- 37 T. Liu, Q.-F. An, Q. Zhao, J.-K. Wu, Y.-H. Song, B.-K. Zhu and C.-J. Gao, *Sci. Rep.*, 2015, **5**, 7782.
- 38 R. A. Ghostine, R. M. Jisr, A. Leahaf and J. B. Schlenoff, *Langmuir*, 2013, **29**, 11742–11750.
- 39 M. Zerball, A. Laschewsky, R. Köhler and R. Von Klitzing, *Polymers*, 2016, **8**, 120.
- 40 R. Steitz, V. Leiner, K. Tauer, V. Khrenov and R. V. Klitzing, *Appl. Phys. A: Mater. Sci. Process.*, 2002, **74**, s519–s521.
- 41 D. M. Reurink, J. P. Haven, I. Achterhuis, S. Lindhoud, H. D. W. Roesink and W. M. de Vos, *Adv. Mater. Interfaces*, 2018, **5**, 1800651.
- 42 G. Decher and J. Schmitt, Darmstadt, 1992.
- 43 G. Decher, *Science*, 1997, **277**, 1232–1237.
- 44 G. Decher, M. Eckle, J. Schmitt and B. Struth, *Curr. Opin. Colloid Interface Sci.*, 1998, **3**, 32–39.
- 45 G. Decher, B. Lehr, K. Lowack, Y. Lvov and J. Schmitt, *Biosens. Bioelectron.*, 1994, **9**, 677–684.
- 46 E. Guzmán, R. G. Rubio and F. Ortega, *Adv. Colloid Interface Sci.*, 2020, **282**, 102197.
- 47 M. M. de Villiers, D. P. Otto, S. J. Strydom and Y. M. Lvov, *Adv. Drug Delivery Rev.*, 2011, **63**, 701–715.
- 48 A. Mateos-Maroto, I. Abelenda-Núñez, F. Ortega, R. G. Rubio and E. Guzmán, *Polymers*, 2021, **13**, 1221.
- 49 A. P. R. Johnston, C. Cortez, A. S. Angelatos and F. Caruso, *Curr. Opin. Colloid Interface Sci.*, 2006, **11**, 203–209.
- 50 K. Ariga, Y. M. Lvov, K. Kawakami, Q. Ji and J. P. Hill, *Adv. Drug Delivery Rev.*, 2011, **63**, 762–771.
- 51 R. M. Iost and F. N. Crespihlo, *Biosens. Bioelectron.*, 2012, **31**, 1–10.
- 52 J. J. Richardson, J. Cui, M. Björnmalm, J. A. Braunger, H. Ejima and F. Caruso, *Chem. Rev.*, 2016, **116**, 14828–14867.
- 53 J. J. Richardson, M. Björnmalm and F. Caruso, *Science*, 2015, **348**, aaa2491.
- 54 M. Michel, V. Toniazzo, D. Ruch and V. Ball, *Int. Scholarly Res. Not.*, 2012, **2012**, 701695.
- 55 G.-R. Xu, S.-H. Wang, H.-L. Zhao, S.-B. Wu, J.-M. Xu, L. Li and X.-Y. Liu, *J. Membr. Sci.*, 2015, **493**, 428–443.
- 56 J. Saqib and I. H. Aljundi, *J. Water Process Eng.*, 2016, **11**, 68–87.



- 57 C. Wang, M. J. Park, H. Yu, H. Matsuyama, E. Drioli and H. K. Shon, *J. Membr. Sci.*, 2022, **661**, 120926.
- 58 N. B. Pramanik and S. L. Regen, *Chem. Rec.*, 2020, **20**, 163–173.
- 59 W. A. Jonkers, E. R. Cornelissen and W. M. de Vos, *J. Membr. Sci.*, 2023, **669**, 121234.
- 60 P. S. Roy, *Ind. Eng. Chem. Res.*, 2024, **63**, 5414–5487.
- 61 L. Shu, Y. Gong, M. Lin, J. Sun and X. Chen, *Appl. Phys. Rev.*, 2024, **11**, 021333.
- 62 S. Chen, Q. Guo and J. Yu, *Aggregate*, 2022, **3**, e293.
- 63 W. C. Blocher and S. L. Perry, *Wiley Interdiscip. Rev.: Nanomed. Nanobiotechnol.*, 2017, **9**, e1442.
- 64 C. A. Quist-Jensen, F. Macedonio and E. Drioli, *Desalination*, 2015, **364**, 17–32.
- 65 B. Krause, M. Storr, T. Ertl, R. Buck, H. Hildwein, R. Deppisch and H. Göhl, *Chem. Ing. Tech.*, 2003, **75**, 1725–1732.
- 66 C. Charcosset, *Food Eng. Rev.*, 2021, **13**, 322–343.
- 67 A. Basile and S. P. Nunes, *Advanced Membrane Science and Technology for Sustainable Energy and Environmental Applications*, 1st edn, 2011.
- 68 K. A. Leamy, S. M. Assmann, D. H. Mathews and P. C. Bevilacqua, *Q. Rev. Biophys.*, 2016, **49**, e10.
- 69 M. Ulbricht, *Polymer*, 2006, **47**, 2217–2262.
- 70 D.-M. Wang and J.-Y. Lai, *Curr. Opin. Chem. Eng.*, 2013, **2**, 229–237.
- 71 G. R. Guillen, Y. Pan, M. Li and E. M. Hoek, *Ind. Eng. Chem. Res.*, 2011, **50**, 3798–3817.
- 72 D. Prat, J. Hayler and A. Wells, *Green Chem.*, 2014, **16**, 4546–4551.
- 73 J. Sherwood, T. J. Farmer and J. H. Clark, *Chem*, 2018, **4**, 2010–2012.
- 74 Q. Zhao, Q. F. An, Y. Ji, J. Qian and C. Gao, *J. Membr. Sci.*, 2011, **379**, 19–45.
- 75 V. S. Meka, M. K. Sing, M. R. Pichika, S. R. Nali, V. R. Kolapalli and P. Kesharwani, *Drug Discovery Today*, 2017, **22**, 1697–1706.
- 76 Q. Zhao, J.-W. Qian, Q.-F. An, Q. Yang and P. Zhang, *J. Membr. Sci.*, 2008, **320**, 8–12.
- 77 Q. Zhao, Y.-L. Ji, J.-K. Wu, L.-L. Shao, Q.-F. An and C.-J. Gao, *RSC Adv.*, 2014, **4**, 52808–52814.
- 78 X.-S. Wang, Y.-L. Ji, P.-Y. Zheng, Q.-F. An, Q. Zhao, K.-R. Lee, J.-W. Qian and C.-J. Gao, *J. Mater. Chem. A*, 2015, **3**, 7296–7303.
- 79 P.-Y. Zheng, W.-H. Zhang, K.-F. Chen, N.-X. Wang and Q.-F. An, *J. Taiwan Inst. Chem. Eng.*, 2019, **95**, 627–634.
- 80 G. Zhumadilova, A. Gazizov, L. Bimendina and S. Kudaibergenov, *Polymer*, 2001, **42**, 2985–2989.
- 81 B. Smitha, S. Sridhar and A. Khan, *Macromolecules*, 2004, **37**, 2233–2239.
- 82 X. Li, C. Liu and B. Van der Bruggen, *J. Mater. Chem. A*, 2020, **8**, 20870–20896.
- 83 E. N. Durmaz, S. Sahin, E. Virga, S. De Beer, L. C. De Smet and W. M. De Vos, *ACS Appl. Polym. Mater.*, 2021, **3**, 4347–4374.
- 84 A. Panagiotopoulos, *J. Phys.: Condens. Matter*, 2009, **21**, 424113.
- 85 J. D. Willott, W. M. Nielen and W. M. de Vos, *ACS Appl. Polym. Mater.*, 2019, **2**, 659–667.
- 86 W. M. Nielen, J. D. Willott and W. M. de Vos, *ACS Appl. Polym. Mater.*, 2020, **2**, 1702–1710.
- 87 W. M. Nielen, J. D. Willott, Z. M. Esguerra and W. M. de Vos, *J. Colloid Interface Sci.*, 2020, **576**, 186–194.
- 88 W. M. Nielen, J. D. Willott, J. A. Galicia and W. M. de Vos, *Polymers*, 2021, **13**, 1775.
- 89 M. I. Baig, E. N. Durmaz, J. D. Willott and W. M. de Vos, *Adv. Funct. Mater.*, 2020, **30**, 1907344.
- 90 M. I. Baig, J. D. Willott and W. M. de Vos, *J. Membr. Sci.*, 2020, **615**, 118502.
- 91 M. I. Baig, P. P. I. Sari, J. Li, J. D. Willott and W. M. de Vos, *J. Membr. Sci.*, 2021, **625**, 119114.
- 92 M. M. H. Mizan, M. Rastgar, S. A. Aktij, A. Asad, P. Karami, A. Rahimpour and M. Sadrzadeh, *J. Membr. Sci.*, 2023, **668**, 121197.
- 93 E. N. Durmaz, J. D. Willott, A. Fatima and W. M. de Vos, *Eur. Polym. J.*, 2020, **139**, 110015.
- 94 M. I. Baig, M. Pejman, J. D. Willott, A. Tiraferri and W. M. De Vos, *ACS Appl. Polym. Mater.*, 2022, **4**, 1010–1020.
- 95 L. Li, M. I. Baig, W. M. de Vos and S. Lindhoud, *ACS Appl. Polym. Mater.*, 2023, **5**, 1810–1818.
- 96 S. Ali and V. M. Prabhu, *Gels*, 2018, **4**, 11.
- 97 S. Lindhoud and M. A. C. Stuart, *Polyelectrolyte complexes in the dispersed and solid state i: Principles and theory*, 2014, pp. 139–172.
- 98 K. Sadman, D. E. Delgado, Y. Won, Q. Wang, K. A. Gray and K. R. Shull, *ACS Appl. Mater. Interfaces*, 2019, **11**, 16018–16026.
- 99 E. N. Durmaz, M. I. Baig, J. D. Willott and W. M. de Vos, *ACS Appl. Polym. Mater.*, 2020, **2**, 2612–2621.
- 100 J. Kamp, S. Emonds, J. Borowec, M. A. R. Toro and M. Wessling, *J. Membr. Sci.*, 2021, **618**, 118632.
- 101 S. Emonds, J. Kamp, J. Borowec, H. Roth and M. Wessling, *Adv. Eng. Mater.*, 2021, **23**, 2001401.
- 102 M. A. Restrepo, S. Emonds, A. Zhao, F. Karakas, J. Kamp, H. Roth and M. Wessling, *J. Membr. Sci.*, 2024, **689**, 122157.
- 103 L. Li, M. I. Baig, W. M. de Vos and S. Lindhoud, *Soft Matter*, 2024, **20**, 5425–5434.
- 104 M. I. Baig, J. D. Willott and W. M. De Vos, *ACS Appl. Polym. Mater.*, 2021, **3**, 3560–3568.
- 105 M. A. Restrepo, J. Kamp, L. Guericke, R. Schnichels, H. Roth and M. Wessling, *J. Membr. Sci. Lett.*, 2023, **3**, 100039.
- 106 J. J. van Lente, M. I. Baig, W. M. de Vos and S. Lindhoud, *J. Colloid Interface Sci.*, 2022, **616**, 903–910.
- 107 *Performance of Bio-based Building Materials*, ed. D. Jones and C. Brischke, Woodhead Publishing, 2017, pp. 187–247, DOI: [10.1016/B978-0-08-100982-6.00004-5](https://doi.org/10.1016/B978-0-08-100982-6.00004-5).
- 108 N. Kampf, *Polym. Adv. Technol.*, 2002, **13**, 895–904.
- 109 V. D. Athawale and R. V. Nimbalkar, *J. Am. Oil Chem. Soc.*, 2011, **88**, 159–185.
- 110 R. R. Smith, A. P. Smith, J. T. Stricker, B. E. Taylor and M. F. Durstock, *Macromolecules*, 2006, **39**, 6071–6074.
- 111 Y. Zhao, J. Hu, X. Hu, F. Zhu, J. Su and J. Han, *Surf. Coat. Technol.*, 2023, **453**, 129143.
- 112 T. Kruk, R. P. Socha, L. Szyk-Warszyńska and P. Warszyński, *Appl. Surf. Sci.*, 2019, **484**, 501–510.



- 113 C. Q. Peng, Y. S. Thio and R. A. Gerhardt, *Nanotechnology*, 2008, **19**, 505603.
- 114 Y. Fu, Y. Zhang and Y. Wang, *Wood Mater. Sci. Eng.*, 2023, **18**, 1553–1561.
- 115 M. Tiitu, J. Laine, R. Serimaa and O. Ikkala, *J. Colloid Interface Sci.*, 2006, **301**, 92–97.
- 116 S. J. Kim, S. G. Yoon, K. B. Lee, Y. D. Park and S. I. Kim, *Solid State Ionics*, 2003, **164**, 199–204.
- 117 J. A. Regenspurg, W. A. Jonkers, M. A. Junker, I. Achterhuis, E. te Brinke and W. M. de Vos, *Desalination*, 2024, **583**, 117693.
- 118 L. M. Petrila, F. Bucatariu, M. Mihai and C. Teodosiu, *Materials*, 2021, **14**, 4152.
- 119 C. Elosua, D. Lopez-Torres, M. Hernaez, I. R. Matias and F. J. Arregui, *Nanoscale Res. Lett.*, 2013, **8**, 539.
- 120 K. C. Krogman, N. S. Zacharia, S. Schroeder and P. T. Hammond, *Langmuir*, 2007, **23**, 3137–3141.
- 121 L. A. Felton, *Int. J. Pharm.*, 2013, **457**, 423–427.
- 122 J. Wang, Y.-F. Xue, X.-C. Chen, M. Hu, K.-F. Ren and J. Ji, *Adv. Healthcare Mater.*, 2020, **9**, 2000381.
- 123 S. Basu, A. Plucinski and J. M. Catchmark, *Green Chem.*, 2017, **19**, 4080–4092.
- 124 K. Chi and J. M. Catchmark, *Food Hydrocolloids*, 2018, **80**, 195–205.
- 125 K. Chi, H. Wang and J. M. Catchmark, *Food Hydrocolloids*, 2020, **103**, 105696.
- 126 M. Haile, O. Sarwar, R. Henderson, R. Smith and J. C. Grunlan, *Macromol. Rapid Commun.*, 2017, **38**, 1600594.
- 127 Y.-H. Yang, M. Haile, Y. T. Park, F. A. Malek and J. C. Grunlan, *Macromolecules*, 2011, **44**, 1450–1459.
- 128 R. J. Smith, C. T. Long and J. C. Grunlan, *Langmuir*, 2018, **34**, 11086–11091.
- 129 H. C. Chiang, T. J. Kolibaba, B. Eberle and J. C. Grunlan, *Macromol. Rapid Commun.*, 2021, **42**, e2000540.
- 130 H.-C. Chiang, B. Eberle, D. Carlton, T. J. Kolibaba and J. C. Grunlan, *ACS Food Sci. Technol.*, 2021, **1**, 495–499.
- 131 R. J. Smith, M. G. Moule, P. A. Leon, E. T. Iverson, T. J. Kolibaba, J. D. Cirillo and J. C. Grunlan, *Macromol. Mater. Eng.*, 2021, **306**, 2100579.
- 132 S. T. Lazar, T. J. Kolibaba and J. C. Grunlan, *Nat. Rev. Mater.*, 2020, **5**, 259–275.
- 133 J. Li, M. H. P. de Heer Kloots, G. van Ewijk, D. J. van Dijken, W. M. de Vos and J. van der Gucht, *Langmuir*, 2024, **40**, 2531–2542.
- 134 A. A. Cain, S. Murray, K. M. Holder, C. R. Nolen and J. C. Grunlan, *Macromol. Mater. Eng.*, 2014, **299**, 1180–1187.
- 135 J. Li, G. van Ewijk, D. J. van Dijken, J. van der Gucht and W. M. de Vos, *ACS Appl. Mater. Interfaces*, 2021, **13**, 21844–21853.
- 136 I. Pietsch, A. Weiss, P. Preishuber-Pflügl, P. Bippus and K. Hünerfauth, *Use of Aqueous Polyanion-Polyethyleneimine Solutions for Producing Polymer Films with Oxygen-Barrier Properties*, Patent EP2859035B1, CN104350091A, EP2859035A1, US20150086734, WO2013182444A1, 2016.
- 137 J. Li, A. Krishna B, G. van Ewijk, D. J. van Dijken, W. M. de Vos and J. van der Gucht, *Colloids Surf., A*, 2022, **648**, 129143.
- 138 M. Hubbe, *BioResources*, 2021, **16**, 4544–4605.
- 139 M. Haile, C. Fincher, S. Fomete and J. C. Grunlan, *Polym. Degrad. Stab.*, 2015, **114**, 60–64.
- 140 H. M. van der Kooij, R. Fokkink, J. van der Gucht and J. Sprakel, *Sci. Rep.*, 2016, **6**, 34383.
- 141 H. M. van der Kooij and J. Sprakel, *Soft Matter*, 2015, **11**, 6353–6359.
- 142 Y. Zhao, Y. Wu, L. Wang, M. Zhang, X. Chen, M. Liu, J. Fan, J. Liu, F. Zhou and Z. Wang, *Nat. Commun.*, 2017, **8**, 2218.
- 143 H. Lee, N. F. Scherer and P. B. Messersmith, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 12999–13003.
- 144 J. H. Waite, *Integr. Comp. Biol.*, 2002, **42**, 1172–1180.
- 145 B. P. Lee, P. B. Messersmith, J. N. Israelachvili and J. H. Waite, *Annu. Rev. Mater. Res.*, 2011, **41**, 99–132.
- 146 R. J. Stewart, T. C. Ransom and V. Hlady, *J. Polym. Sci., Part B: Polym. Phys.*, 2011, **49**, 757–771.
- 147 M. J. Brennan, B. F. Kilbride, J. J. Wilker and J. C. Liu, *Biomaterials*, 2017, **124**, 116–125.
- 148 J. D. White and J. J. Wilker, *Macromolecules*, 2011, **44**, 5085–5088.
- 149 Y. Zhou, C. Zhang, S. Gao, W. Li, J.-J. Kai and Z. Wang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 50451–50460.
- 150 C. Heinzmann, C. Weder and L. M. de Espinosa, *Chem. Soc. Rev.*, 2016, **45**, 342–358.
- 151 H. Lee, S. M. Dellatore, W. M. Miller and P. B. Messersmith, *Science*, 2007, **318**, 426–430.
- 152 H. Lee, B. P. Lee and P. B. Messersmith, *Nature*, 2007, **448**, 338–341.
- 153 L. Xue, B. Sanz, A. Luo, K. T. Turner, X. Wang, D. Tan, R. Zhang, H. Du, M. Steinhart, C. Mijangos, M. Guttman, M. Kappl and A. del Campo, *ACS Nano*, 2017, **11**, 9711–9719.
- 154 M. Follador, F. Tramacere and B. Mazzolai, *Bioinspiration Biomimetics*, 2014, **9**, 046002.
- 155 P. G. Lawrence and Y. Lapitsky, *Langmuir*, 2015, **31**, 1564–1574.
- 156 Q. Peng, J. Chen, T. Wang, L. Gong, X. Peng, M. Wu, Y. Ma, F. Wu, D. Yang, H. Zhang and H. Zeng, *J. Mater. Chem. A*, 2021, **9**, 12988–13000.
- 157 R. J. Stewart, C. S. Wang and H. Shao, *Adv. Colloid Interface Sci.*, 2011, **167**, 85–93.
- 158 S. Kim, H. Y. Yoo, J. Huang, Y. Lee, S. Park, Y. Park, S. Jin, Y. M. Jung, H. Zeng, D. S. Hwang and Y. Jho, *ACS Nano*, 2017, **11**, 6764–6772.
- 159 J.-S. Jhiang, T.-H. Wu, C.-J. Chou, Y. Chang and C.-J. Huang, *J. Mater. Chem. B*, 2019, **7**, 2878–2887.
- 160 B. K. Ahn, S. Das, R. Linstadt, Y. Kaufman, N. R. Martinez-Rodriguez, R. Mirshafian, E. Kesselman, Y. Talmon, B. H. Lipshutz, J. N. Israelachvili and J. H. Waite, *Nat. Commun.*, 2015, **6**, 8663.
- 161 W. Wei, Y. Tan, N. R. Martinez Rodriguez, J. Yu, J. N. Israelachvili and J. H. Waite, *Acta Biomater.*, 2014, **10**, 1663–1670.
- 162 I. Kaminker, W. Wei, A. M. Schrader, Y. Talmon, M. T. Valentine, J. N. Israelachvili, J. H. Waite and S. Han, *Soft Matter*, 2017, **13**, 9122–9131.
- 163 H. Shao, K. N. Bachus and R. J. Stewart, *Macromol. Biosci.*, 2009, **9**, 464–471.



- 164 D. S. Hwang, H. Zeng, A. Srivastava, D. V. Krogstad, M. Tirrell, J. N. Israelachvili and J. H. Waite, *Soft Matter*, 2010, **6**, 3232–3236.
- 165 Y. Huang, P. G. Lawrence and Y. Lapitsky, *Langmuir*, 2014, **30**, 7771–7777.
- 166 Z. Bashir, W. Yu, Z. Xu, Y. Li, J. Lai, Y. Li, Y. Cao and B. Xue, *Int. J. Mol. Sci.*, 2022, **23**, 9987.
- 167 W. Sun, T. Liu, X. Zhang, X. Zhang, Q. Yan, J. Yin and S. Luan, *Adv. Healthcare Mater.*, 2023, **12**, 2300669.
- 168 C. Cui and W. Liu, *Prog. Polym. Sci.*, 2021, **116**, 101388.
- 169 A. Narayanan, A. Dhinojwala and A. Joy, *Chem. Soc. Rev.*, 2021, **50**, 13321–13345.
- 170 H. Fan and J. P. Gong, *Adv. Mater.*, 2021, **33**, 2102983.
- 171 M. Vahdati, D. Hourdet and C. Creton, *Prog. Polym. Sci.*, 2023, **139**, 101649.
- 172 M. Li, A. Mao, Q. Guan and E. Saiz, *Chem. Soc. Rev.*, 2024, **53**, 8240–8305.
- 173 Y. Lu, X. Xu and J. Li, *J. Mater. Chem. B*, 2023, **11**, 3338–3355.
- 174 J.-N. Zhang, H. Zhu, T. Liu, Y. Chen, C. Jiao, C. He and H. Wang, *Polymer*, 2020, **206**, 122845.
- 175 M. Vahdati, F. J. Cedano-Serrano, C. Creton and D. Hourdet, *ACS Appl. Polym. Mater.*, 2020, **2**, 3397–3410.
- 176 M. Dompé, F. J. Cedano-Serrano, O. Heckert, N. van den Heuvel, J. van der Gucht, Y. Tran, D. Hourdet, C. Creton and M. Kamperman, *Adv. Mater.*, 2019, **31**, 1808179.
- 177 L. van Westerveld, T. Pelras, A. H. Hofman, K. Loos, M. Kamperman and J. Es Sayed, *Macromolecules*, 2024, **57**, 652–663.
- 178 M. Dompé, F. J. Cedano-Serrano, M. Vahdati, L. van Westerveld, D. Hourdet, C. Creton, J. van der Gucht, T. Kodger and M. Kamperman, *Adv. Mater. Interfaces*, 2020, **7**, 1901785.
- 179 I. A. van Hees, A. H. Hofman, M. Dompé, J. van der Gucht and M. Kamperman, *Eur. Polym. J.*, 2020, **141**, 110034.
- 180 L. van Westerveld, J. Es Sayed, M. de Graaf, A. H. Hofman, M. Kamperman and D. Parisi, *Soft Matter*, 2023, **19**, 8832–8848.
- 181 H. Shao and R. J. Stewart, *Adv. Mater.*, 2010, **22**, 729–733.
- 182 X. Feng, K. Pouw, V. Leung and R. Pelton, *Biomacromolecules*, 2007, **8**, 2161–2166.
- 183 C. Wei, X. Zhu, H. Peng, J. Chen, F. Zhang and Q. Zhao, *ACS Sustainable Chem. Eng.*, 2019, **7**, 4508–4514.
- 184 X. Li, L. Shang, D. Li, W. Wang, S. Chen, H. Zhong, Y. Huang and S. Long, *Polym. Test.*, 2022, **109**, 107547.
- 185 R. Qie, S. Zajforoushan Moghaddam and E. Thormann, *ACS Sustainable Chem. Eng.*, 2024, **12**, 4456–4463.
- 186 G. Hernandez-Pérez, G. Goma and J. L. Rols, *Water Res.*, 1999, **33**, 1837–1844.
- 187 K. Ushimaru, T. Morita and T. Fukuoka, *J. Wood Chem. Technol.*, 2020, **40**, 172–177.
- 188 I. Aranaz, A. R. Alcántara, M. C. Civera, C. Arias, B. Elorza, A. Heras Caballero and N. Acosta, *Polymers*, 2021, **13**, 3256.
- 189 Q. Zhao, D. W. Lee, B. K. Ahn, S. Seo, Y. Kaufman, J. N. Israelachvili and J. H. Waite, *Nat. Mater.*, 2016, **15**, 407–412.
- 190 C. Wei, C. Zhao, Y. Ru, Y. Sun, T. Zhao, S. Qi, J. Zhou and M. Liu, *Adv. Funct. Mater.*, 2024, 2400581.
- 191 P. Galland, M. H. Iqbal, D. Favier, M. Legros, P. Schaaf, F. Boulmedais and M. Vahdati, *J. Colloid Interface Sci.*, 2024, **661**, 196–206.
- 192 X. Liu, M. Haddou, I. Grillo, Z. Mana, J.-P. Chapel and C. Schatz, *Soft Matter*, 2016, **12**, 9030–9038.
- 193 S. Kim, J. Huang, Y. Lee, S. Dutta, H. Y. Yoo, Y. M. Jung, Y. Jho, H. Zeng and D. S. Hwang, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, E847–E853.
- 194 J. Zhang, H. Lei, M. Qin, W. Wang and Y. Cao, *Supramol. Mater.*, 2022, **1**, 100005.
- 195 H. Fan, J. Wang, Z. Tao, J. Huang, P. Rao, T. Kurokawa and J. P. Gong, *Nat. Commun.*, 2019, **10**, 5127.
- 196 M. Cui, S. Ren, S. Wei, C. Sun and C. Zhong, *APL Mater.*, 2017, **5**, 116102.
- 197 Y. Chen, J. Meng, Z. Gu, X. Wan, L. Jiang and S. Wang, *Adv. Funct. Mater.*, 2020, **30**, 1905287.
- 198 H. Y. Yuen, H. P. Bei and X. Zhao, *Chem. Eng. J.*, 2022, **431**, 133372.
- 199 Q. F. Wang and J. B. Schlenoff, *Macromolecules*, 2014, **47**, 3108–3116.
- 200 C. H. Porcel and J. B. Schlenoff, *Biomacromolecules*, 2009, **10**, 2968–2975.
- 201 H. H. Hariri and J. B. Schlenoff, *Macromolecules*, 2010, **43**, 8656–8663.
- 202 R. Steitz, W. Jaeger and R. von Klitzing, *Langmuir*, 2001, **17**, 4471–4474.
- 203 K. Glinel, A. Moussa, A. M. Jonas and A. Laschewsky, *Langmuir*, 2002, **18**, 1408–1412.
- 204 B. Vishalakshi, S. Ghosh and V. Kalpagam, *Polymer*, 1993, **34**, 3270–3275.
- 205 A. Shakya, M. Girard, J. T. King and M. O. de la Cruz, *Macromolecules*, 2020, **53**, 1258–1269.
- 206 C. Schatz, J. M. Lucas, C. Viton, A. Domard, C. Pichot and T. Delair, *Langmuir*, 2004, **20**, 7766–7778.
- 207 Z. J. Sui, D. Salloum and J. B. Schlenoff, *Langmuir*, 2003, **19**, 2491–2495.
- 208 J. Yang, M. Y. Shen, T. Wu, Y. Luo, M. Y. Li, H. L. Wen and J. H. Xie, *Food Chem.*, 2020, **333**, 127493.
- 209 A. S. Michaels, L. Mir and N. S. Schneide, *J. Phys. Chem.*, 1965, **69**, 1447–1455.
- 210 Y. H. Chen, M. Yang, S. Abou Shaheen and J. B. Schlenoff, *Macromolecules*, 2021, **54**, 7890–7899.
- 211 A. Krishna, J. D. Willott, S. Lindhoud and W. M. De Vos, *Polymer*, 2022, **242**, 124583.
- 212 F. B. Zhu, L. B. Cheng, J. Yin, Z. L. Wu, J. Qian, J. Z. Fu and Q. Zheng, *ACS Appl. Mater. Interfaces*, 2016, **8**, 31304–31310.
- 213 J. K. Bediako, J. H. Kang, Y. S. Yun and S. H. Choi, *ACS Appl. Polym. Mater.*, 2022, **4**, 2346–2354.
- 214 J. Akintola, Z. A. Digby and J. B. Schlenoff, *ACS Appl. Mater. Interfaces*, 2023, **15**, 9962–9969.
- 215 J. K. Bediako, E. S. M. Mouele, Y. El Ouardi and E. Repo, *Chem. Eng. J.*, 2023, **462**, 142322.
- 216 A. S. Michaels, *Ind. Eng. Chem.*, 1965, **57**, 32–40.
- 217 L. Gubler, *Curr. Opin. Electrochem.*, 2019, **18**, 31–36.
- 218 M. Ahmad and M. Ahmed, *J. Appl. Electrochem.*, 2023, **53**, 1537–1562.



- 219 H. Strathmann, A. Grabowski and G. Eigenberger, *Ind. Eng. Chem. Res.*, 2013, **52**, 10364–10379.
- 220 Y. X. Yang, P. Li, X. B. Zheng, W. P. Sun, S. X. Dou, T. Y. Ma and H. G. Pan, *Chem. Soc. Rev.*, 2022, **51**, 9620–9693.
- 221 J. Ran, L. Wu, Y. B. He, Z. J. Yang, Y. M. Wang, C. X. Jiang, L. Ge, E. Bakangura and T. W. Xu, *J. Membr. Sci.*, 2017, **522**, 267–291.
- 222 S. A. N. Mehrabani, V. Vatanpour and I. Koyuncu, *Sep. Purif. Technol.*, 2022, **298**, 121691.
- 223 J. Rozière and D. J. Jones, *Annu. Rev. Mater. Res.*, 2003, **33**, 503–555.
- 224 B. A. Krishna, S. Lindhoud and W. M. de Vos, *J. Colloid Interface Sci.*, 2021, **593**, 11–20.
- 225 B. A. Krishna, W. M. de Vos and S. Lindhoud, *Langmuir*, 2023, **39**, 6874–6884.
- 226 A. Krishna, H. J. Zwijnenberg, S. Lindhoud and W. M. de Vos, *J. Membr. Sci.*, 2022, **652**, 120463.
- 227 M. B. Arif, S. Kheawhom and S. T. Dubas, *J. Energy Storage*, 2023, **74**, 109425.
- 228 P. Luangaramvej and S. T. Dubas, *e-Polym.*, 2021, **21**, 194–199.
- 229 M. Yang, Z. A. Digby, Y. Chen and J. B. Schlenoff, *Sci. Adv.*, 2022, **8**, eabm4783.
- 230 D. Kitto and J. Kamcev, *J. Membr. Sci.*, 2023, **677**, 121608.
- 231 M. Yang, Z. A. Digby and J. B. Schlenoff, *Macromolecules*, 2020, **53**, 5465–5474.
- 232 H. M. Fares, Q. F. Wang, M. Yang and J. B. Schlenoff, *Macromolecules*, 2019, **52**, 610–619.
- 233 R. F. Shamoun, H. H. Hariri, R. A. Ghostine and J. B. Schlenoff, *Macromolecules*, 2012, **45**, 9759–9767.
- 234 Y. S. Ye and Y. A. Elabd, *Macromolecules*, 2011, **44**, 8494–8503.
- 235 S. Chempath, B. R. Einsla, L. R. Pratt, C. S. Macomber, J. M. Boncella, J. A. Rau and B. S. Pivovar, *J. Phys. Chem. C*, 2008, **112**, 3179–3182.
- 236 M. G. Marino and K. D. Kreuer, *ChemSusChem*, 2015, **8**, 513–523.
- 237 Y. X. Liu, M. H. Zhu, Q. Zhao, Q. F. An, J. W. Qian, K. Lee and J. Y. Lai, *J. Membr. Sci.*, 2011, **385**, 132–140.
- 238 D. M. Reurink, J. D. Willott, H. D. W. Roesink and W. M. de Vos, *ACS Appl. Polym. Mater.*, 2020, **2**, 5278–5289.
- 239 M. A. Restrepo, S. Emonds, A. Zhao, F. Karakas, J. Kamp, H. Roth and M. Wessling, *J. Membr. Sci.*, 2024, **689**, 122157.
- 240 J. A. Asenjo and B. A. Andrews, *J. Chromatogr. A*, 2011, **1218**, 8826–8835.
- 241 M. Iqbal, Y. Tao, S. Xie, Y. Zhu, D. Chen, X. Wang, L. Huang, D. Peng, A. Sattar, M. A. B. Shabbir, H. I. Hussain, S. Ahmed and Z. Yuan, *Biol. Proced. Online*, 2016, **18**, 18.
- 242 A. L. Grilo, M. Raquel Aires-Barros and A. M. Azevedo, *Sep. Purif. Rev.*, 2016, **45**, 68–80.
- 243 S. Boeynaems, S. Alberti, N. L. Fawzi, T. Mittag, M. Polymenidou, F. Rousseau, J. Schymkowitz, J. Shorter, B. Wolozin, L. Van Den Bosch, P. Tompa and M. Fuxreiter, *Trends Cell Biol.*, 2018, **28**, 420–435.
- 244 E. Gomes and J. Shorter, *J. Biol. Chem.*, 2019, **294**, 7115–7127.
- 245 M. Chen, R. Xu, Y. Wu, J. Xiong, S. Z. Keleş and N. P. Hankins, *J. Water Process Eng.*, 2024, **64**, 105528.
- 246 A. A. Tokmakov, A. Kurotani and K. I. Sato, *Front. Mol. Biosci.*, 2021, **8**, 775736.
- 247 I. Rayment, in *Encyclopedia of Physical Science and Technology*, ed. R. A. Meyers, Academic Press, New York, 3rd edn, 2003, pp. 191–218, DOI: [10.1016/B0-12-227410-5/00616-5](https://doi.org/10.1016/B0-12-227410-5/00616-5).
- 248 S. B. Ebrahimi and D. Samanta, *Nat. Commun.*, 2023, **14**, 2411.
- 249 T. Tang, N. Wu, S. Tang, N. Xiao, Y. Jiang, Y. Tu and M. Xu, *J. Agric. Food Chem.*, 2023, **71**, 1788–1801.
- 250 M. Zhao, S. A. Eghtesadi, M. B. Dawadi, C. Wang, S. Huang, A. E. Seymore, B. D. Vogt, D. A. Modarelli, T. Liu and N. S. Zacharia, *Macromolecules*, 2017, **50**, 3818–3830.
- 251 A. Shah and N. I. Malek, *J. Ionic Liq.*, 2021, **1**, 100006.
- 252 L. Li, S. Srivastava, M. Andreev, A. B. Marciel, J. J. de Pablo and M. V. Tirrell, *Macromolecules*, 2018, **51**, 2988–2995.
- 253 A. E. Neitzel, Y. N. Fang, B. Yu, A. M. Romyantsev, J. J. de Pablo and M. V. Tirrell, *Macromolecules*, 2021, **54**, 6878–6890.
- 254 R. Staño, J. J. van Lente, S. Lindhoud and P. Košovan, *Macromolecules*, 2024, **57**, 1383–1398.
- 255 W. C. Blocher McTigue and S. L. Perry, *Soft Matter*, 2019, **15**, 3089–3103.
- 256 W. C. Blocher McTigue and S. L. Perry, *Small*, 2020, **16**, 1907671.
- 257 K. A. Black, D. Priftis, S. L. Perry, J. Yip, W. Y. Byun and M. Tirrell, *ACS Macro Lett.*, 2014, **3**, 1088–1091.
- 258 S. Lindhoud and M. M. A. E. Claessens, *Soft Matter*, 2016, **12**, 408–413.
- 259 J. J. van Lente, M. M. A. E. Claessens and S. Lindhoud, *Biomacromolecules*, 2019, **20**, 3696–3703.
- 260 J. J. van Lente and S. Lindhoud, *Small*, 2022, **18**, 2105147.
- 261 J. van Lente, M. Pazos Urrea, T. Brouwer, B. Schuur and S. Lindhoud, *Green Chem.*, 2021, **23**, 5812–5824.
- 262 X. Mi, W. C. Blocher McTigue, P. U. Joshi, M. K. Bunker, C. L. Heldt and S. L. Perry, *Biomater. Sci.*, 2020, **8**, 7082–7092.
- 263 P. U. Joshi, C. Decker, X. Zeng, A. Sathyavageswaran, S. L. Perry and C. L. Heldt, *Biomacromolecules*, 2024, **25**, 741–753.
- 264 C. H. Porcel and J. B. Schlenoff, *Biomacromolecules*, 2009, **10**, 2968–2975.
- 265 M. Müller, L. Wirth and B. Urban, *Appl. Spectrosc.*, 2024, **78**, 56–66.
- 266 G. Ferrand-Drake del Castillo, R. L. N. Hailes and A. Dahlin, *J. Phys. Chem. Lett.*, 2020, **11**, 5212–5218.
- 267 S. M. Lalwani, P. Batys, M. Sammalkorpi and J. L. Lutkenhaus, *Macromolecules*, 2021, **54**, 7765–7776.
- 268 Y. Zhang, F. Li, L. D. Valenzuela, M. Sammalkorpi and J. L. Lutkenhaus, *Macromolecules*, 2016, **49**, 7563–7570.
- 269 Z. A. Digby, M. Yang, S. Lteif and J. B. Schlenoff, *Macromolecules*, 2022, **55**, 978–988.
- 270 U. Scheler, *Curr. Opin. Colloid Interface Sci.*, 2009, **14**, 212–215.
- 271 S. Akoka, L. Barantin and M. Trierweiler, *Anal. Chem.*, 1999, **71**, 2554–2557.



- 272 G. Allegri, J. Huskens, R. P. Martinho and S. Lindhoud, *J. Colloid Interface Sci.*, 2024, **672**, 654–663.
- 273 L. Barantin, A. L. Pape and S. Akoka, *Magn. Reson. Med.*, 1997, **38**, 179–182.
- 274 R. P. Choudhury and M. Schönhoff, *J. Chem. Phys.*, 2007, **127**, 234702.
- 275 L. Patel, O. Mansour, M. Crossman and P. Griffiths, *Langmuir*, 2019, **35**, 9233–9238.
- 276 N. Proietti, M. E. Amato, G. Masci and A. L. Segre, *Macromolecules*, 2002, **35**, 4365–4372.
- 277 A. B. Ruiz-Muelle, P. G. Moreno and I. Fernández, *Talanta*, 2021, **230**, 122344.
- 278 S. M. Lalwani, K. Hellikson, P. Batys and J. L. Lutkenhaus, *Macromolecules*, 2024, **57**, 4695–4705.
- 279 V. Guillou and T. Schönberger, *J. Pharm. Biomed. Anal.*, 2022, **213**, 114690.
- 280 T. Minato and M. Satoh, *J. Polym. Sci., Part B: Polym. Phys.*, 2004, **42**, 4412–4420.
- 281 S. A. Watson, A. J. Edwards and J. A. Parkinson, *Anal. Lett.*, 2017, **50**, 161–172.
- 282 J. L. Ochoa, S. Germann, B. Conklin, K. Kurita, D. J. Russell, C. Yang and J. G. Napolitano, *Magn. Reson. Chem.*, 2024, **62**, 4–10.
- 283 T. V. Chalikian, M. Totrov, R. Abagyan and K. J. Breslauer, *J. Mol. Biol.*, 1996, **260**, 588–603.
- 284 H. Jiang, P. Taranekar, J. R. Reynolds and K. S. Schanze, *Angew. Chem., Int. Ed.*, 2009, **48**, 4300–4316.
- 285 B. J. Olson and J. Markwell, *Curr Protoc Protein Sci*, 2007, ch. 3, Unit 3.4.
- 286 A. C. Obermeyer, C. E. Mills, X. H. Dong, R. J. Flores and B. D. Olsen, *Soft Matter*, 2016, **12**, 3570–3581.
- 287 C. S. Cummings and A. C. Obermeyer, *Biochemistry*, 2018, **57**, 314–323.
- 288 V. Scandella, R. C. Paolicelli and M. Knobloch, *Sci. Rep.*, 2020, **10**, 14642.
- 289 M. R. Soboleski, J. Oaks and W. P. Halford, *FASEB J.*, 2005, **19**, 440–442.
- 290 E. Spruijt, A. H. Westphal, J. W. Borst, M. A. Cohen Stuart and J. van der Gucht, *Macromolecules*, 2010, **43**, 6476–6484.

