



Cite this: *Nanoscale Horiz.*, 2019, 4, 975

Received 7th January 2019,  
Accepted 5th March 2019

DOI: 10.1039/c9nh00006b

rsc.li/nanoscale-horizons

## Nanoscale antiadhesion properties of sophorolipid-coated surfaces against pathogenic bacteria†

Claire Valotteau,<sup>a</sup> Niki Baccile,<sup>b</sup> Vincent Humblot,<sup>c</sup> Sophie Roelants,<sup>de</sup> Wim Soetaert,<sup>d</sup> Christian V. Stevens<sup>b,f</sup> and Yves F. Dufrêne<sup>b,\*g</sup>

A current challenge in nanomedicine is to develop innovative strategies to fight infections caused by multiresistant bacterial pathogens. A striking example is antiadhesion therapy, which represents an attractive alternative to antibiotics to prevent and treat biofilm-associated infections on medical devices. By means of single-cell force nanoscopy, we demonstrate that sophorolipid (SL) biosurfactants feature unusually strong antiadhesion properties against *Staphylococcus aureus* and *Escherichia coli*, two nosocomial pathogens involved in catheter-related infections, which represent a major public health problem worldwide. We find that the nanoscale adhesion forces of single bacteria are much weaker on SL monolayers than on abiotic alkanethiol monolayers. The remarkable antifouling efficacy of SL-surfaces is likely to involve repulsive hydration forces associated with sophorose head-groups. We also show that, owing to their surfactant properties, soluble SLs block bacterial adhesion forces towards abiotic surfaces. Collectively, our single-cell experiments demonstrate that sophorolipids exhibit strong and versatile antiadhesion properties, making them promising candidates to design anti-infective biomaterials.

### Conceptual insights

Many bacterial pathogens attach to biomaterials such as catheters, and grow to form biofilms. These surface-associated multicellular communities cause infections that are difficult to eradicate. Antiadhesion therapy represents an attractive alternative to antibiotics to prevent and treat biofilm infections. Using single-cell experiments, we show that glycolipids feature unusual high-efficiency and broad-range antiadhesive properties against Gram-positive and Gram-negative pathogens, making them ideally suited to fight biomaterial-associated infections. Our study shows that glycosylated substrates, classically used to favor protein and cell interactions, exhibit strong antifouling properties that could be used to design anti-infective biomaterials. We anticipate that force nanoscopy will become an important tool in nanomedicine to test the efficiency of bioinspired antifouling compounds.

## Introduction

Many bacterial pathogens attach to biomaterials such as catheters, and grow to form biofilms.<sup>1,2</sup> These surface-associated multicellular

communities cause infections that are difficult to treat because bacteria within biofilms are inherently more resistant to antibiotics. Bacterial adhesion to medical materials involves a wealth of molecular interactions, that are either specific (adhesin–ligand interactions) or non-specific (hydrophobic and electrostatic interactions).<sup>3</sup>

An attractive strategy to fight biomaterial-related infections is the use of surface coatings with anti-infective properties, relying either on biocidal or antiadhesive activities.<sup>4–7</sup> The advantage of antiadhesion surfaces is that they do not involve the spread of antibacterial compounds in the environment, therefore limiting the development of resistant strains. Effective antiadhesion coatings should be able to inhibit both specific (e.g. lectin-sugar) and non-specific (e.g. hydrophobic) adhesive interactions. End-grafted poly(ethylene glycol) (PEG) chains represent a popular strategy for the design of antifouling biomaterials.<sup>7,8</sup> An interesting alternative is the use of carbohydrate compounds, which feature antiadhesion properties, low toxicity and can be produced in large amounts.<sup>9,10</sup> Among these, microbial glycolipids are biosurfactants<sup>11–13</sup> well-known for their antimicrobial properties,<sup>14</sup> which involve both biocidal<sup>15,16</sup> and antiadhesive effects.<sup>15,17–19</sup>

Sophorolipids (SLs) are glycolipids featuring strong anti-microbial action,<sup>20</sup> yet the underlying molecular mechanism

<sup>a</sup> Louvain Institute of Biomolecular Science and Technology, Université catholique de Louvain, Croix du Sud, 4-5, bte L7.07.06., B-1348 Louvain-la-Neuve, Belgium. E-mail: Yves.Dufrene@uclouvain.be

<sup>b</sup> Sorbonne Université, Centre National de la Recherche Scientifique, Laboratoire de Chimie de la Matière Condensée de Paris, LCMCP, F-75005 Paris, France

<sup>c</sup> Sorbonne Université, Centre National de la Recherche Scientifique, Laboratoire de Réactivité de Surface, LRS, F-75005 Paris, France

<sup>d</sup> Ghent University, Centre for Industrial Biotechnology and Biocatalysis (InBio.be), Coupure Links 653, B-9000 Gent, Belgium

<sup>e</sup> Bio Base Europe Pilot Plant, Rodenhuiszekaai 1, B-9000 Gent, Belgium

<sup>f</sup> SynBioC Research Group, Department of Green Chemistry and Technology, Faculty of Bioscience Engineering, Ghent University, Coupure links 653, B-9000 Gent, Belgium

<sup>g</sup> Walloon Excellence in Life sciences and Biotechnology (WELBIO), B-1300 Wavre, Belgium

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9nh00006b





**Fig. 1** Probing the antiadhesion activity of sophorolipids on the nanoscale. (a) Single-cell force spectroscopy was used to quantify the nanoscale interactions between Gram-positive (*S. aureus*) and Gram-negative (*E. coli*) bacterial pathogens, and substrates coated with monolayers of alkanethiols (hydrophobic, in red), hydrophilic, in blue) or sophorolipids (SLs, in orange). (b) We compared the properties of several SL-derivatives: sophorolipids (SLs), sophorose-free oleic acid (OA), SL-derivative with a saturated aliphatic chain ( $SL_0$ ). (c) The ability of soluble SLs to inhibit bacterial adhesion forces towards alkanethiol monolayers was also studied. The cartoons are not to scale.

remain controversial.<sup>15,16,21–23</sup> Using single-cell force nanoscopy,<sup>24,25</sup> we measure the nanoscale adhesive forces between single bacteria and monolayers of alkanethiols or deacetylated C18:1 acidic, where the carboxyl group has been modified to allow binding onto gold surfaces (Fig. 1a).<sup>16,22</sup> We focus on the Gram-positive bacterium *Staphylococcus aureus*, a nosocomial pathogen frequently implicated in the colonization of biomaterials such as central venous catheters and prosthetic joints,<sup>26</sup> as well as on the uropathogenic Gram-negative *Escherichia coli*, known to be commonly involved in catheter-associated urinary tract infections.<sup>5</sup> We compare the passivating activity of SL-surfaces with that of surfaces modified with sophorose-free oleic acid (OA) which mimics only the aliphatic chain of sophorolipids or with a C18:0 SL-derivative having a fully saturated aliphatic chain ( $SL_0$ ) (Fig. 1b). We also show the ability of soluble

SLs to inhibit bacterial adhesion forces towards alkanethiol monolayers (Fig. 1c). Our results are of clinical significance as they open up new avenues for the design of infection-resistant biomaterials.

## Results and discussion

### Sophorolipid-surfaces show strong antiadhesion activity against *S. aureus*

We first sought to compare the nanoscale adhesive interactions between *S. aureus* and alkanethiol or SL monolayers (Fig. 2). By means of single-cell force spectroscopy,<sup>27,28</sup> we investigated the forces between single Newman wild-type (WT) bacteria and hydrophobic methyl-terminated monolayers (water contact angle  $\sim 110^\circ$ ).<sup>29</sup> Hydrophobic surfaces were used because hydrophobic interactions represent one of the driving forces for the adhesion of bacterial pathogens to biomaterials.<sup>30</sup> Fig. 2a (left panel) shows the adhesion forces measured between single *S. aureus* bacteria and hydrophobic monolayers (three replicates; for data on more cells see Fig. 2b and c, red filled bars). Essentially all force curves (adhesion probability of  $98 \pm 2\%$ ; mean  $\pm$  s.d. from 13 cells) showed large adhesion events of  $3220 \pm 2067$  pN magnitude (mean  $\pm$  s.d.,  $n = 2825$  adhesive curves from 13 cells). Adhesion profiles generally featured multiple sequential peaks (inset in Fig. 2a, left panel) with long extensions ( $258 \pm 187$  nm,  $n = 2825$ ), suggesting that cell detachment lead to the stretching of large cell surface molecules. We postulated that these adhesion signatures are associated with the binding and unfolding of hydrophobic domains from cell surface proteins. To test this hypothesis, we studied the adhesion of a mutant deficient in sortase A,<sup>31</sup> the enzyme responsible for anchoring surface proteins to peptidoglycan (hereafter referred to as Newman  $\Delta srtA$ ). As can be seen in Fig. 2b and c (red open bars) and Fig. S1a (ESI<sup>†</sup>) (red histogram), the interaction between Newman  $\Delta srtA$  cells and hydrophobic surfaces lead to much lower adhesion probability ( $54 \pm 34\%$ , 8 cells) and adhesion forces ( $119 \pm 64$  pN,  $n = 489$  adhesive curves from 8 cells), suggesting that the strong forces of Newman WT bacteria are due to protein-dependent hydrophobic adhesion. Supporting this notion, adhesion between Newman WT cells and hydrophilic hydroxyl-terminated monolayers (water contact angle  $\sim 30^\circ$ )<sup>29</sup> (Fig. 2a, middle panel in blue, and Fig. 2c, blue filled bars) is much weaker (adhesion forces of  $226 \pm 185$  pN,  $n = 2021$  adhesive curves from 13 cells) than on hydrophobic surfaces, and only featured single adhesive events. Even weaker adhesive forces ( $92 \pm 22$  pN,  $n = 320$  curves from 7 cells) were observed with Newman  $\Delta srtA$  cells (Fig. 2c and Fig. S1, ESI<sup>†</sup>, blue), suggesting the involvement of surface proteins in hydrophilic adhesion. These results show that bacterial cell surface proteins are engaged in strong hydrophobic interactions with surfaces, while they also modestly contribute to hydrophilic adhesion. We speculate that, under force, hydrophobic protein domains are unfolded and exposed, thereby promoting hydrophobic adhesion. The  $\sim 3000$  pN forces are much larger than those generally associated with single proteins at similar





**Fig. 2** Nanoscale adhesion forces of the nosocomial pathogen *Staphylococcus aureus* are strongly inhibited on SL-coated surfaces. (a) Adhesion force histograms with representative force curves recorded in buffer between three *S. aureus* Newman wild-type (WT) cells and hydrophobic methyl- (red), hydrophilic hydroxyl- (blue) and SL- (orange) terminated monolayers. (b and c) Adhesion probability (% of adhesive curves) and adhesion forces of the *S. aureus* Newman WT (filled bars) vs. the Newman  $\Delta$ srtA mutant (open bars) strains. Box-charts show the mean values (squares), the median, the 25% and 75% quartiles (boxes), and the standard deviation (whiskers).

loading rates ( $\sim 50$ – $200$  pN),<sup>32</sup> suggesting they reflect the binding and unfolding of a large number of adhesins. The  $\sim 250$  nm extensions correspond to elongated polypeptide chains of  $\sim 700$  amino acids, which is in the range expected for staphylococcal adhesins.<sup>26</sup> Besides, we note that for both WT and  $\Delta$ srtA strains, the adhesion probability on hydrophilic surfaces was substantial ( $48 \pm 38\%$  and  $43 \pm 43\%$  from 13 WT and 7  $\Delta$ srtA cells, respectively), pointing to the involvement of glycopolymers such as teichoic acids and peptidoglycan in hydrophilic interactions.

Analysis of WT bacterial adhesion forces on SL monolayers (water contact angle  $\sim 50^\circ$ )<sup>22</sup> (Fig. 2a, right panel; Fig. 2b and c, orange filled bars) revealed adhesion forces of  $249 \pm 149$  pN ( $n = 1186$  adhesive curves from 10 cells) much lower than on hydrophobic surfaces (Fig. 2b and c, red filled bars), and in the range of those on hydrophilic surfaces (Fig. 2b and c, blue filled bars). In addition, the binding probability of the Newman WT strain was lower on SL surfaces ( $36 \pm 35\%$ ) than on hydrophilic surfaces (Fig. 2b). The Newman  $\Delta$ srtA strain showed even lower binding probability (Fig. 2b and Fig. S1a, ESI†, orange), which suggests that SL targets not only surface proteins but also, to some extent, other components like teichoic acids and peptidoglycan. Overall, SL-surfaces display weaker adhesion than hydrophobic and hydrophilic surfaces,

indicating that the structure of the glycolipid is critical for its antiadhesion activity.

### Sophorolipid-surfaces show a broad spectrum of antiadhesion properties

During infection, the Gram-negative uropathogenic *E. coli* bacterium specifically binds to urinary epithelial cells using FimH, a widely-investigated mannose-binding protein expressed on cell surface pili. Whether FimH also contributes to non-specific adhesion to abiotic surfaces is unclear. Interaction forces between cells overexpressing FimH (*E. coli* FimH<sup>+</sup>) and hydrophobic monolayers (Fig. 3a, left panel) featured adhesion events of  $774 \pm 724$  pN magnitude ( $n = 2956$  adhesive curves from 20 cells; Fig. 3c, red filled bar), which, although substantial, are clearly weaker than for *S. aureus* (Fig. 2). Also, mostly single adhesion peaks were observed (inset in Fig. 3a, left panel), leading us to believe that the molecular origin of hydrophobic binding differed in the two species. The nanoscale adhesive properties of *E. coli* cells which poorly express FimH (*E. coli* FimH<sup>−</sup>, Fig. S1b, ESI†) were similar to those of *E. coli* FimH<sup>+</sup> cells (Fig. 3a), meaning FimH was not engaged in hydrophobic adhesion (Fig. 3b and c, filled vs. open red bars). Adhesion of both FimH<sup>+</sup> and FimH<sup>−</sup> strains was almost abrogated on hydrophilic monolayers ( $17 \pm 12\%$  and  $14 \pm 10\%$ , from 20 FimH<sup>+</sup> and





**Fig. 3** Nanoscale adhesion of uropathogenic *Escherichia coli* is also reduced on SL-coated surfaces. (a) Adhesion force histograms with representative force curves recorded in buffer between three *E. coli* FimH<sup>+</sup> cells and hydrophobic methyl- (red), hydrophilic hydroxyl- (blue) and SL- (orange) terminated monolayers. (b and c) Adhesion probability (% of adhesive curves) and adhesion forces of the *E. coli* FimH<sup>+</sup> (filled bars) vs. *E. coli* FimH<sup>-</sup> (open bars) strains. Box-charts show the mean values (squares), the median, the 25% and 75% quartiles (boxes), and the standard deviation (whiskers).

13 FimH<sup>-</sup> cells, Fig. 3b filled vs. open blue bars), confirming that strong adhesion is mostly hydrophobic in nature. Of note, rupture distances of *E. coli* FimH<sup>+</sup> cells were rather short on all substrates ( $379 \pm 430$  nm and  $247 \pm 317$  nm,  $n = 3863$  and 1154 adhesive curves from 20 cells, for hydrophobic and hydrophilic surfaces, respectively). This is in sharp contrast with the interaction between *E. coli* FimH<sup>+</sup> cells and mannose-coated surface, which showed elongated force plateaus (up to  $\sim 5000$  nm in length) reflecting pili unfolding.<sup>33</sup> All together, our observations show that FimH is not engaged in non-specific adhesion to surfaces, pointing to the role of other surface components such as outer membrane proteins and lipopolysaccharides.

Notably, SL-surfaces featured very weak adhesion forces (Fig. 3a, right panel), without any substantial differences between the FimH<sup>+</sup> ( $192 \pm 114$  pN,  $n = 2059$  adhesive curves from 20 cells) and FimH<sup>-</sup> ( $169 \pm 84$  pN,  $n = 1868$  adhesives curves from 13 cells) strains (Fig. 3c, orange bars). The binding probability appeared similar for both strains ( $47 \pm 28\%$  and  $59 \pm 27\%$ , Fig. 3b, orange bars). Hence, while FimH interacts with mannosyl derivatives, it does not specifically bind to sophorose units. Similarly, others sugar-binding proteins such as Concanavalin A were also unable to bind sophorose.<sup>34,35</sup>

Collectively, these single-cell experiments demonstrate that SL-coated surfaces exhibit a broad spectrum of antiadhesion activity, being able to prevent the adhesion of both Gram-positive and Gram-negative pathogens. SLs interfere with *S. aureus*

adhesion primarily by blocking protein-dependent hydrophobic forces, and to some extent by limiting glycopolymer interactions



**Fig. 4** Role of the sugar group and of the aliphatic chain in antiadhesion. Adhesion probabilities (a and c) and adhesion forces (b and d) of *S. aureus* (a and b) Newman WT (filled bars) and  $\Delta$ srtA (open bars) cells and *E. coli* (c and d) FimH<sup>+</sup> (filled bars) and FimH<sup>-</sup> (open bars) cells on surfaces modified with SL (orange), sophorose-free oleic acid (OA, green) and a SL-derivative with a saturated aliphatic chain (SL<sub>0</sub>, burgundy). Box-charts show the mean values (squares), the median, the 25% and 75% quartiles (boxes), and the standard deviation (whiskers).





via e.g. teichoic acids and peptidoglycan. For *E. coli*, SL surfaces are insensitive to carbohydrate-binding protein FimH, and largely inhibit non-specific hydrophobic forces.

### Molecular origin of the passivating effect of sophorolipids

As SLs are amphiphilic molecules, we wondered whether the structure of the carbohydrate group and of the aliphatic chain play important roles in their antiadhesive behavior. To test this, we probed bacterial adhesion forces towards substrates modified with sophorose-free oleic acid (OA) and with a fully saturated

aliphatic chain C18:0 sophorolipid-derivative (SL<sub>0</sub>) (Fig. 4; water contact angles of SL, SL<sub>0</sub> and OA monolayers:  $48 \pm 2^\circ$ ,  $71 \pm 2^\circ$  and  $55 \pm 2^\circ$ ).<sup>22</sup> For all strains, the adhesion frequency was much higher on OA-surfaces ( $68 \pm 40\%$ ,  $59 \pm 30\%$ ,  $75 \pm 33\%$ , and  $84 \pm 25\%$  from 10 *S. aureus* WT, 8 *AsrtA* cells, 17 *E. coli* FimH<sup>+</sup> and 20 FimH<sup>-</sup> cells, respectively; Fig. 4a and c) than on SL-surfaces ( $36 \pm 35\%$ ,  $14 \pm 8\%$ ,  $47 \pm 28\%$  and  $58 \pm 27\%$  from 10, 8, 20 and 13 cells, respectively; Fig. 4a and c), implying that the sophorose group plays an important role in preventing adhesion. Increased, yet less pronounced, adhesion frequency was also observed on



**Fig. 5** Soluble sophorolipids inhibit bacterial adhesion to abiotic surfaces. (a) Variation of the adhesion probability (a and c) and adhesion forces (b and d) measured between three representative cells (square, round and triangle symbols) of *S. aureus* Newman WT (a and b) or *E. coli* FimH<sup>+</sup> (c and d), and hydrophobic methyl- (a and c in red) or hydrophilic hydroxyl- (b and d in blue) terminated monolayers, upon addition of increasing concentrations of sophorolipids (SLs).





Bio Base Europe Pilot Plant (Batch 5 in ref. 45). Deacetylated acidic fully saturated C18 sophorolipids (SL<sub>0</sub>) was derived from the previous one using a Pd-catalyzed hydrogenation reaction described in ref. 46.

Sophorolipids (SL, SL<sub>0</sub>) and oleic acid (OA) were grafted onto flat polycrystalline gold substrates *via* a self-assembled monolayer of short aminothiols, as described previously.<sup>22</sup> Briefly, glass coverslips coated with a thin layer of gold by vacuum evaporation were immersed in an ethanol solution containing 10 mM of cysteamine (cys) for 3 h, rinsed with ethanol, and dried with N<sub>2</sub>. Substrates and cantilevers were then immersed for 3 h in a solution containing 50 mg mL<sup>-1</sup> sophorolipids whose carboxylic acid termination having been previously activated into a succinimide ester using a mixture of EDC and NHS (molar ratio NHS:EDC = 1:2). Finally, after incubation, sonication and successive rinsing in ultrapure water and ethanol were carried out to remove non-covalently grafted reactants before drying under a flow of N<sub>2</sub>. These coatings were previously characterized by infrared spectroscopy, X-ray photoelectron spectroscopy and quartz-crystal microbalance.<sup>16,22</sup> All these techniques confirm that the glycolipids were efficiently attached to the surface *via* amide bonds, forming a layer of around 30 Å.

Hydrophobic and hydrophilic substrates were prepared by immersing gold-coated substrates in ethanol solutions containing 1 mM 1-dodecanethiol (Sigma-Aldrich; 98%) or 1 mM 11-mercapto-1-undecanol (Sigma-Aldrich; 97%) overnight, rinsing them with ethanol, and drying them under N<sub>2</sub>.

### Bacterial strains and growth conditions

Wild-type (WT) *S. aureus* strain Newman and its mutant ( $\Delta$ *srtA*) deficient in sortase A, the enzyme responsible for anchoring CWA proteins to peptidoglycan, were cultured in Trypticase soy broth (TSB) overnight at 37 °C under agitation, then washed twice in prewarmed TSB and grown again at 37 °C to exponential phase (until an optical density at 600 nm of 0.3 was reached). *E. coli* strains UTI89 WT (FimH<sup>-</sup>) and the FimH UTI89 strain engineered for continuous fimbriation (FimH<sup>+</sup>) were cultivated overnight in lysogeny broth (LB) at 37 °C under agitation. Before AFM experiments, cells were harvested by centrifugation, washed twice in phosphate-buffered saline (PBS), and diluted 1:100 in PBS.

### Single-cell force spectroscopy

Colloidal probes were obtained by attaching a single silica microsphere (6.1 µm in diameter; Bangs Laboratories) with a thin layer of UV-curable glue (NOA 63; Norland Edmund Optics) onto triangular tipless cantilevers (NP-O10; Bruker) and using a NanoWizard III AFM System (JPK Instruments). These colloidal probes were immersed for 1 h in Tris-buffered saline (pH 8.5) containing 4 mg mL<sup>-1</sup> dopamine hydrochloride (Sigma), rinsed in Tris-buffered saline (pH 8.5), and used directly for cell-probe preparation. The nominal spring constant of the colloidal probe cantilever was ~0.06 N m<sup>-1</sup>, as determined by the thermal noise method.

For cell probe preparation, 50 µL of a diluted cell suspension was deposited into a Petri dish where the functionalized surfaces were glued on double tape at a distinct location, and the system was filled with 3 mL of PBS. The colloidal probe was brought into contact with an isolated bacterium and retracted to attach the bacterial cell; proper attachment of the cell on the colloidal probe was checked using optical microscopy. The cell probe was then moved towards the surfaces to measure cell-surface interactions. Force curves were recorded at room temperature using an applied force of 0.25 nN, a constant approach-retraction speed of 1.0 µm s<sup>-1</sup>, and a contact time of 100 ms. For inhibition with soluble SLs experiments, the sophorolipids were added at increasing concentrations from 1 µg mL<sup>-1</sup> up to 100 µg mL<sup>-1</sup>. Data were analyzed using the Data Processing software from JPK Instruments. Adhesion force and distance rupture histograms were obtained by calculating the maximum adhesion force and rupture distance of the last peak for each curve. The adhesion probability is calculated as the percentage of curves displaying adhesive events over the total number of recorded curves.

### Author contributions

C. V., N. B., V. H. and Y. F. D. designed the experiments, analyzed the data and wrote the article. C. V. collected the data. S. R., C. S. and W. S. produced and purified the glycolipids. All authors discussed the results and commented the article.

### Conflicts of interest

The authors declare no competing financial interest.

### Acknowledgements

Work at the Université catholique de Louvain was supported by the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation programme (grant agreement No [693630]), the National Fund for Scientific Research (FNRS), the FNRS-WELBIO (Grant no. WELBIO-CR-2015A-05), and the Research Department of the Communauté française de Belgique (Concerted Research Action). Y. F. D. is a Research Director at the FNRS. We thank Joan Geoghegan and Stéphane Vincent for providing us with bacterial strains.

### References

- 1 J. W. Costerton, P. S. Stewart and E. P. Greenberg, *Science*, 1999, **284**, 1318–1322.
- 2 R. Kolter and E. P. Greenberg, *Nature*, 2006, **441**, 300–302.
- 3 H. J. Busscher, W. Norde and H. C. Van Der Mei, *Appl. Environ. Microbiol.*, 2008, **74**, 2559–2564.
- 4 D. Campoccia, L. Montanaro and C. R. Arciola, *Biomaterials*, 2013, **34**, 8533–8554.
- 5 D. M. Siddiq and R. O. Darouiche, *Nat. Rev. Urol.*, 2012, **9**, 305–314.



- 6 H. J. Busscher, H. C. van der Mei, G. Subbiahdoss, P. C. Jutte, J. J. van den Dungen, S. A. Zaat, M. J. Schultz and D. W. Grainger, *Sci. Transl. Med.*, 2012, **4**, 153rv110.
- 7 I. Francolini, C. Vuotto, A. Piozzi and G. Donelli, *APMIS*, 2017, **125**, 392–417.
- 8 D. L. Elbert and J. A. Hubbell, *Annu. Rev. Mater. Sci.*, 1996, **26**, 365–394.
- 9 M. Morra, *Biomacromolecules*, 2005, **6**, 1205–1223.
- 10 V. Gadenne, L. Lebrun, T. Jouenne and P. Thebault, *Colloids Surf., B*, 2013, **112**, 229–236.
- 11 S. S. Cameotra and R. S. Makkar, *Curr. Opin. Microbiol.*, 2004, **7**, 262–266.
- 12 P. Singh and S. S. Cameotra, *Trends Biotechnol.*, 2004, **22**, 142–146.
- 13 L. R. Rodrigues and J. A. Teixeira, *Adv. Exp. Med. Biol.*, 2010, **672**, 75–87.
- 14 M. Ines and G. Dhouha, *Carbohydr. Res.*, 2015, **416**, 59–69.
- 15 A. d. J. Cortés-Sánchez, H. Hernández-Sánchez and M. E. Jaramillo-Flores, *Microbiol. Res.*, 2013, **168**, 22–32.
- 16 C. Valotteau, I. M. Banat, C. A. Mitchell, H. Lydon, R. Marchant, F. Babonneau, C.-M. Pradier, N. Baccile and V. Humblot, *Colloids Surf., B*, 2017, **157**, 325–334.
- 17 I. Ofek, D. L. Hasty and N. Sharon, *FEMS Immunol. Med. Microbiol.*, 2003, **38**, 181–191.
- 18 S. Kesel, A. Mader, P. Seeberger, O. Lieleg and M. Opitz, *Appl. Environ. Microbiol.*, 2014, **80**, 5911–5917.
- 19 A. L. Flores-Mireles, J. N. Walker, M. Caparon and S. J. Hultgren, *Nat. Rev. Microbiol.*, 2015, **13**, 269.
- 20 E. I. P. Delbeke, M. Movsisyan, K. M. Van Geem and C. V. Stevens, *Green Chem.*, 2016, **18**, 76–104.
- 21 M. A. Díaz De Rienzo, P. Stevenson, R. Marchant and I. M. Banat, *FEMS Microbiol. Lett.*, 2016, **363**, fnv224.
- 22 C. Valotteau, C. Calers, S. Casale, J. Berton, C. V. Stevens, F. Babonneau, C.-M. Pradier, V. Humblot and N. Baccile, *ACS Appl. Mater. Interfaces*, 2015, **7**, 18086–18095.
- 23 C. Pontes, M. Alves, C. Santos, M. H. Ribeiro, L. Gonçalves, A. F. Bettencourt and I. A. Ribeiro, *Int. J. Pharm.*, 2016, **513**, 697–708.
- 24 J. Xiao and Y. F. Dufrene, *Nat. Microbiol.*, 2016, **1**, 16186.
- 25 Y. F. Dufrene, *ACS Nano*, 2017, **11**, 19–22.
- 26 T. J. Foster, J. A. Geoghegan, V. K. Ganesh and M. Hook, *Nat. Rev. Microbiol.*, 2014, **12**, 49–62.
- 27 A. Beaussart, S. El-Kirat-Chatel, R. M. A. Sullan, D. Alsteens, P. Herman, S. Derclaye and Y. F. Dufrène, *Nat. Protoc.*, 2014, **9**, 1049–1055.
- 28 A. Beaussart, S. El-Kirat-Chatel, P. Herman, D. Alsteens, J. Mahillon, P. Hols and Y. F. Dufrène, *Biophys. J.*, 2013, **104**, 1886–1892.
- 29 D. Alsteens, E. Dague, P. G. Rouxhet, A. R. Baulard and Y. F. Dufrène, *Langmuir*, 2007, **23**, 11977–11979.
- 30 R. J. Doyle, *Microbes Infect.*, 2000, **2**, 391–400.
- 31 S. K. Mazmanian, G. Liu, E. R. Jensen, E. Lenoy and O. Schneewind, *Proc. Natl. Acad. Sci. U. S. A.*, 2000, **97**, 5510–5515.
- 32 D. J. Muller, J. Helenius, D. Alsteens and Y. F. Dufrene, *Nat. Chem. Biol.*, 2009, **5**, 383–390.
- 33 A. Beaussart, M. Abellán-Flos, S. El-Kirat-Chatel, S. P. Vincent and Y. F. Dufrène, *Nano Lett.*, 2016, **16**, 1299–1307.
- 34 I. J. Goldstein, R. N. Iyer, E. E. Smith and L. L. So, *Biochemistry*, 1967, **6**, 2373–2377.
- 35 R. N. Iyer and I. J. Goldstein, *Immunochemistry*, 1973, **10**, 313–322.
- 36 V. A. Parsegian and T. Zemb, *Curr. Opin. Colloid Interface Sci.*, 2011, **16**, 618–624.
- 37 M. Kanduc, A. Schlaich, A. H. de Vries, J. Jouhet, E. Marechal, B. Deme, R. R. Netz and E. Schneck, *Nat. Commun.*, 2017, **8**, 14899.
- 38 K. L. Prime and G. M. Whitesides, *J. Am. Chem. Soc.*, 1993, **115**, 10714–10721.
- 39 K. Prime and G. Whitesides, *Science*, 1991, **252**, 1164–1167.
- 40 A. M. Krachler and K. Orth, *Virulence*, 2013, **4**, 284–294.
- 41 J. A. Geoghegan, T. J. Foster, P. Speziale and Y. F. Dufrene, *Trends Microbiol.*, 2017, **25**, 512–514.
- 42 C. N. Spaulding, R. D. Klein, S. Ruer, A. L. Kau, H. L. Schreiber, Z. T. Cusumano, K. W. Dodson, J. S. Pinkner, D. H. Fremont, J. W. Janetka, H. Remaut, J. I. Gordon and S. J. Hultgren, *Nature*, 2017, **546**, 528–532.
- 43 L. K. Mahal, *Chem. Biol.*, 2004, **11**, 1602–1604.
- 44 N. Laurent, J. Voglmeir and S. L. Flitsch, *Chem. Commun.*, 2008, 4400–4412, DOI: 10.1039/B806983M.
- 45 P. Dhasaiyan, P. Le Griel, S. Roelants, E. Redant, I. N. Van Bogaert, S. Prevost, B. L. Prasad and N. Baccile, *Chem-PhysChem*, 2017, **18**, 643–652.
- 46 A. S. Cuvier, J. Berton, C. V. Stevens, G. C. Fadda, F. Babonneau, I. N. Van Bogaert, W. Soetaert, G. Pehau-Arnaudet and N. Baccile, *Soft Matter*, 2014, **10**, 3950–3959.

