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Direct heteroarylation polymerization: guidelines for defect-free conjugated polymers†

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Direct (hetero)arylation polymerization (DHAP) has emerged as a valuable and atom-economical alternative to traditional cross-coupling methods for the synthesis of low-cost and efficient conjugated polymers for organic electronics. However, when applied to the synthesis of certain (hetero)arene-based materials, a lack of C–H bond selectivity has been observed. To prevent such undesirable side-reactions, we report the design and synthesis of new, bulky, phosphine-based ligands that significantly enhance selectivity of the DHAP process for both halogenated and non-halogenated electron-rich and electron-deficient thiophene-based comonomers. To better understand the selectivity issues, density functional theory (DFT) calculations have been performed on various halogenated and non-halogenated electron-rich and electron-deficient thiophene-based comonomers. Calculations showed that the presence of bromine atoms decreases the energy of activation (E_a) of the adjacent C–H bonds, allowing undesirable β -defects for some brominated aromatic units. Both calculations and the new ligands should lead to the rational design of monomers and methods for the preparation of defect-free conjugated polymers from DHAP.

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1. Introduction

Conjugated polymers are the object of considerable attention from both academic and industrial laboratories. They often combine the excellent electrical and optical properties of metals and inorganic semiconducting materials with the mechanical flexibility, simple processing, and low production cost of synthetic polymers. Their good solubility facilitates large-scale processing techniques such as roll-to-roll printing, which makes low-cost printed electronic device fabrication possible. Polymer solar cells (PSCs)^{1–4} and organic field-effect transistors (OFETs)^{5–9} are among the most promising applications for conjugated polymers. With power conversion efficiencies having surpassed 10%,^{10–12} lifetimes of nearly 15 years^{13,14} and hole mobilities up to $36 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$,¹⁵ conjugated polymers

seem to have now reached the requirements for commercial applications.^{16,17}

The development of plastic electronics relies on innovative, robust and versatile coupling methods, such as Ziegler–Natta,^{18,19} Migita–Stille,²⁰ Kumada,²¹ Heck,²² Miyaura–Suzuki,²³ Negishi²⁴ and olefin-metathesis^{25–27} to produce well-defined conjugated polymers. However, production cost, presence of impurities, batch-to-batch variations and industrial scalability are still major issues for the implementation of plastic electronics. For instance, Migita–Stille cross-coupling polymerization techniques, which involve the use of organostannanes (synthesized *via* multistep organometallic processes that require demanding purification steps), lead to stoichiometric quantities of toxic by-products such as Me_3SnBr . Clearly, cheaper and more atom-efficient methodologies are highly desirable for the preparation of semiconducting organic polymers and their large-scale applications.

For these reasons, direct (hetero)arylation polymerization (DHAP) is generating great interest among the polymer science community and is now considered an important asset for polymer chemists despite the early stage of its development.^{28–34} Tracing its origins to the synthesis of small organic molecules by direct (hetero)arylation, this polymerization method facilitates C–C bond formation between (hetero)arenes and (hetero)aryl halides (most commonly I and Br) catalyzed by palladium complexes. Unlike the Migita–Stille cross-coupling polymerization, only benign by-products are generated during the DHAP (mainly CsHCO_3 and CsBr). Up to now, this polymerization

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method has been successfully applied to the synthesis of many conjugated polymers based on key monomers such as 3-alkylthiophene,^{35–39} benzodithiophene (BDT),^{40–42} 5-alkylthieno [3,4-*c*]pyrrole-4,6-dione (TPD)^{43–46} and 3,6-bis(thiophen-2-yl)-2,5-bis(2-alkyl)pyrrolo[3,4-*c*]pyrrole-1,4-dione (DPP)^{47–54} derivatives.

Despite recent achievements, significant challenges, mainly for some thiophene-based derivatives that feature several active C–H bonds, remain to be overcome, namely: (1) α - α homocouplings side-reactions and (2) regioselectivity issues (α , β

activated hydrogen atoms) that lead to β -defects (see Fig. 1). Mechanistically speaking, direct (hetero)arylation involves the oxidative addition of an (hetero)arylhalide substrate to a palladium catalyst followed by the heterolytic cleavage of a C–H bond of a second (hetero)aryl substrate, leading to the formation of a new C–C bond *via* reductive elimination. Most (hetero)arenes undergo a concerted metalation-deprotonation (CMD) process in which the co-ligand directly participates in the proton transfer.³⁰ Presently, a limited number of phosphine-based ligands yield high reactivity and selectivity for DHAP. Among them, $P(o\text{-NMe}_2(\text{C}_6\text{H}_4))_3$ has proven itself to be suitable to the preparation of highly regioregular poly(alkylthiophene)s while $P(o\text{-OMe}(\text{C}_6\text{H}_4))_3$ seems adequate for the copolymerization of bromoarenes with heteroarenes.⁵⁵ Inspired by the pioneering work done by Buchwald *et al.*⁵⁶ on the design of specific phosphine ligands (SPhos, XPhos, DavePhos, *etc.*) for palladium-catalyzed formation of C–C, C–N and C–O bonds and the studies by Thompson *et al.*,⁵⁷ Bura *et al.*³² and Dubnik *et al.*⁵⁸ demonstrating the positive influence of bulky carboxylic acid additives on the DHAP selectivity, we report new phosphine-based ligands closely related to $P(o\text{-OMe}(\text{C}_6\text{H}_4))_3$ (**L_{REF}**, Fig. 2). The phosphines were made bulkier at the *ortho*-position (**L1–L5**) using a low cost, simple and efficient methodology in the hope that they will enhance DHAP selectivity. The effect of the phosphine-based ligands was investigated for the polymerization of various brominated and non-brominated electron-rich

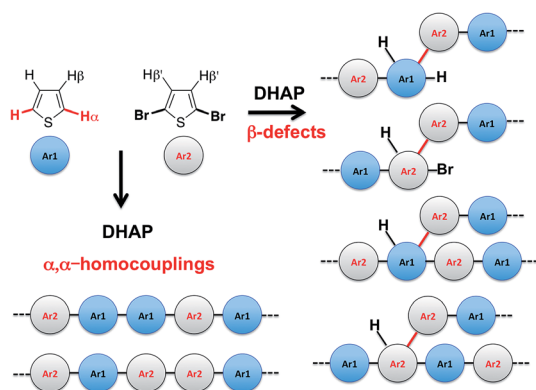


Fig. 1 Most probable structural defects upon direct (hetero)arylation polymerization (DHAP).

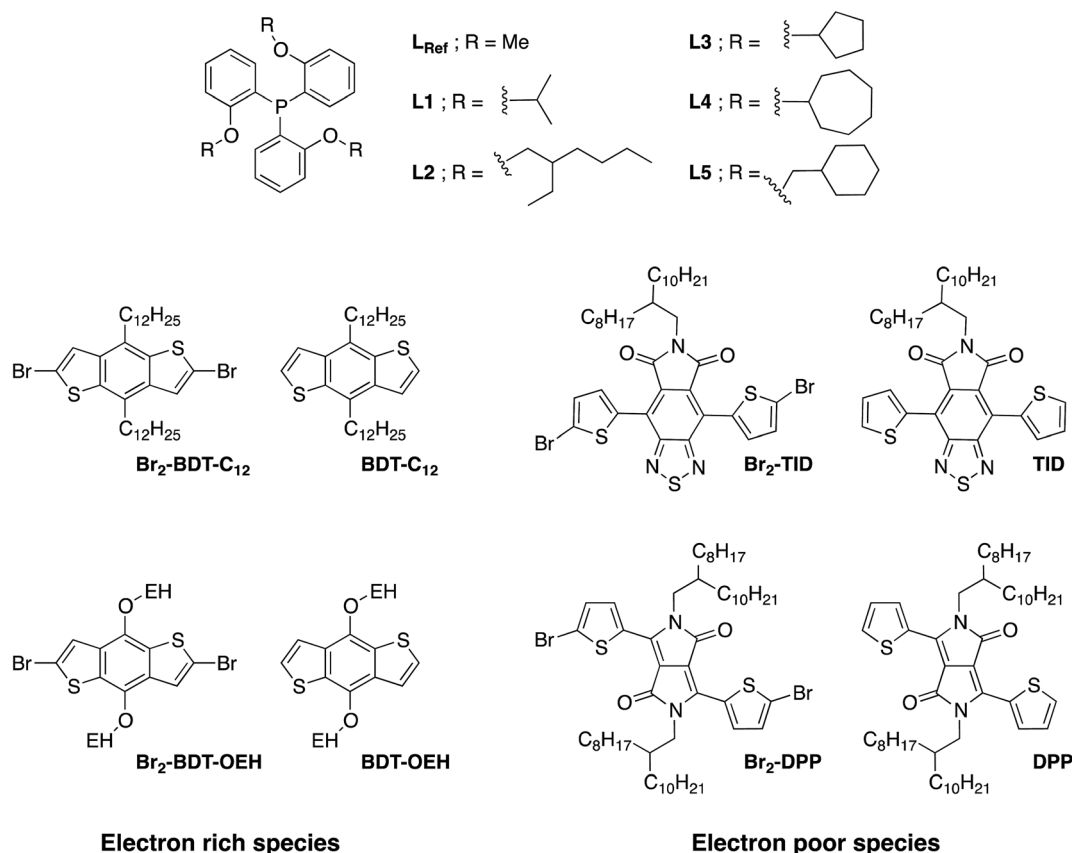


Fig. 2 The chemical structures of the phosphines and comonomers studied in this work.

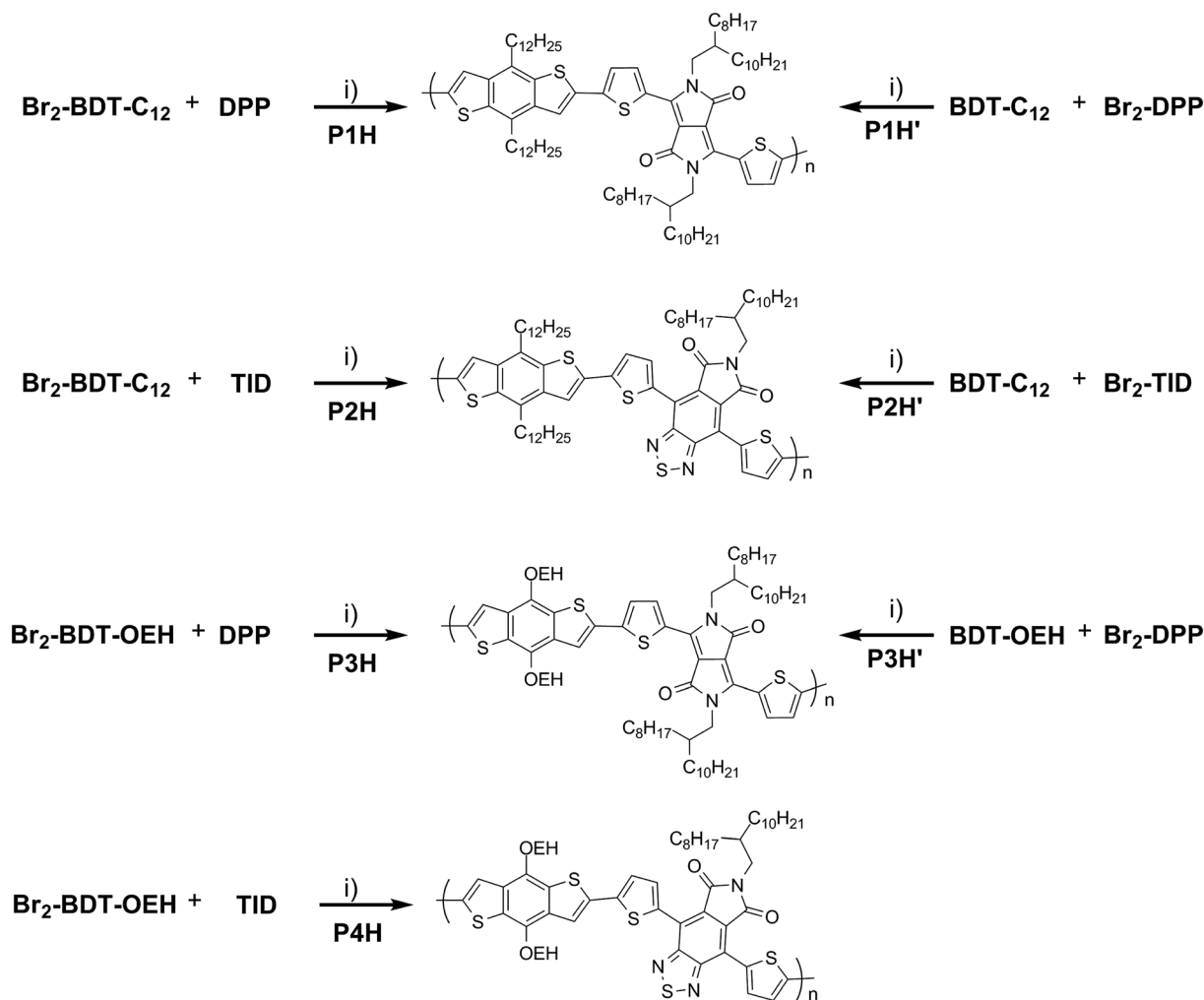


and electron-poor thiophene-based monomers (Fig. 2). The influence of the nature of the substituents was also analyzed. Finally, density functional theory (DFT) calculations were used to rationalize the selectivity issues observed for some halogenated and non-halogenated electron-rich and electron-deficient benzo[1,2-*b*:4,5-*b'*]dithiophene (BDT), pyrrolo[3,4-*c*]-pyrrole-1,4-dione (DPP) and thiadiazolo[3,4-*e*]isoindole-5,7-dione (TID) derivatives. DFT calculations were also utilized to explain why some well-defined and nearly defect-free conjugated polymers reported in literature were successfully prepared by DHAP.

2. Results and discussion

As reported in the ESI,[†] the reference copolymers were first synthesized using the well-established Migita–Stille cross-coupling polymerization using the following synthetic procedure: Pd₂dba₃/ligand (ratio 1 : 4) and equimolar amounts of comonomers in toluene at 110 °C led to polymers **P1S** to **P4S**. As shown in Table S2,[†] high number-average molecular weight

($\overline{M}_n$) of 56 and 57 kg mol^{−1} was obtained for **P1S** and **P3S** when triphenyl arsine (AsPh₃) was used as ligand for the Stille cross-coupling polymerization.⁵⁹ Unlike **P1S** and **P3S**, the same catalytic system led to **P2S** and **P4S** with lower molecular weights. The molecular weight of **P2S** is limited due to its poor solubility (linear side chain on BDT). Indeed, regardless of the polymerization conditions, precipitation of the growing polymer chain occurs within few minutes. On the other hand, high molecular weight **P4S** (alkoxy side chain on BDT) can be obtained when P(*o*-Tolyl)₃ is used as ligand instead of AsPh₃. Indeed, $\overline{M}_n$ up to 150 kg mol^{−1} was reached, evidencing both the better efficiency of Pd₂dba₃/P(*o*-Tolyl)₃ catalytic system and the effect of the side chain on the of the BDT unit (alkoxy *vs.* alkyl). However the lack of solubility of the higher molecular weight batches led to broad and featureless ¹H NMR spectra. For comparison purposes in terms of molecular weights, Pd₂dba₃/AsPh₃ catalytic system was therefore used as reference. All polymers (soluble fraction after Soxhlet extractions) obtained by Migita–Stille cross-coupling polymerization are soluble in chlorinated solvents such as



Scheme 1 Synthetic routes for **P1H** to **P4H** and **P1H'** to **P3H'** by DHAP.



We first evaluated the polymerization of Br₂-BDT-C₁₂ (a halogenated electron-rich thiophene-based derivative with chemically reactive β-C-H bonds) with DPP (a non-halogenated electron-poor thiophene-based derivative with reactive α and β C-H bonds), two highly-studied and efficient building blocks in organic electronics (Scheme 1). γ-C-H bonds on DPP are inactive to C-H activation due to steric hindrance.^{47,61} The copolymerization of these units yielded **PIH**. On the basis of previous studies^{32,62} and as a reference point for comparison purposes, we kept the following polymerization conditions constant throughout this study: to a 0.5 mol L⁻¹ solution of the monomers in toluene were added 5 mol% of Pd(OAc)₂, 20 mol% of phosphine, 3 equivalents of Cs₂CO₃, and 1 equivalent of pivalic acid (PivOH). The mixture was reacted at 125 °C (see Experimental details in ESI†). Polymerization using tris(*o*-methoxyphenyl)phosphine (**L_{REF}** in Fig. 2) was used as a control because of its generally satisfactory catalytic activity and selectivity in the DHAP (see ESI† for initial screening of DHAP conditions, Table S1†). The polymerization data are reported in Table S3† and the UV-Vis absorption spectra (in solution) of the resulting materials are shown in Fig. 3. For each ligand (**L1**–**L5**),

Figure 1 consists of two line graphs, (a) and (b), showing the UV-Vis absorption spectra of various polymers. Both graphs plot Absorbance (a.u.) on the y-axis (ranging from 0 to 1) against wavelength (nm) on the x-axis (ranging from 350 to 1000 nm).

Graph (a) shows the spectra for P1S (red line), P1H-Lref (black line), P1H-L1 (blue line), and P1H-L2 (green line). All four curves exhibit a broad absorption band peaking around 750 nm, with absorbance values reaching approximately 0.9-1.0. There is also a smaller shoulder or peak around 400 nm.

Graph (b) shows the spectra for P1S (red line), P1H-L3 (black line), P1H-L4 (blue line), and P1H-L5 (green line). These curves also show a broad absorption band peaking around 750 nm, with absorbance values reaching approximately 0.9-1.0. Similar to graph (a), there is a smaller shoulder or peak around 400 nm.

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In the absence of well-defined NMR data, organic field-effect transistors (OFETs) (see ESI†) were utilized to probe electrical properties of polymers. Indeed, while the performance of such devices depend on tedious processing optimization steps and while OFETs measurements will not allow the accurate identification or quantification of the defects (β -defects or α - α homocouplings), the charge mobility in the devices is still expected to be affected by defects in the polymer (Table 1). In this study, **P1S** was used as a benchmark (even though the presence of DPP–DPP homocoupling within the main chain is suspected) and was compared to **P1H-L_{REF}**, **P1H-L3** and **P1H-L4**. All polymers showed ambipolar behavior. As reported in Table 1, the OFET results are in good agreement with the UV-Vis absorption data for the **P1H** polymer series. Based on the above, **P1H-L_{REF}** (low molecular weight and different UV-Vis absorption spectra as compared to **P1S**) is the least efficient material. On the other hand, both **P1H-L3** and **P1H-L4** (higher molecular weights and displaying a good overlap of the UV-Vis absorption spectra without the shoulder at 820 nm) led to average hole and electron mobilities at least as good as than those measured for **P1S** and better than those of **P1H-L_{REF}**. It is worth noting that the average hole mobilities obtained for the **P1H** series are in the same range as those reported by Morse *et al.*⁶⁵ All these results suggest that bulky phosphine ligands result in well-defined

To validate this hypothesis, we then applied this catalytic system to the synthesis of a copolymer containing a promising electron-deficient moiety studied in organic electronics: bis(thiophen-2-yl)-6-(2-octyldodecyl)-[1,2,5]thiadiazolo[3,4-*c*]isoindole-5,7-dione (TID) (Fig. 2).⁶⁶⁻⁶⁸ The experimental conditions in this study were the same as those used for the **P1H** series. The polymerization data are summarized in Table S4† and the UV-Vis absorption spectra are shown in Fig. 4. The effect of the bulkiness of the phosphine for **P2H** was the same as **P1H**. Both polymerization times and yields (around 75%) are similar for most of the ligands except for **P2H-L_{REF}** where the yield of the soluble fraction was reduced to 39%. As observed for the **P1H** series, the bulkiest phosphine (**L4**) led to the highest number-average molecular weight (**P2H-L4**, 21 kg mol⁻¹). Fig. 4 shows that the UV-Vis absorption spectra of both **P2H-L_{REF}** and **P2H-L1** are quite different from that of **P2S**. While the band gaps ($E_g = 1.48$ eV) are similar, the vibronic bands of the polymer prepared by the different polymerization methods do not overlap one another. Although **P2H-L_{REF}** and **P2H-L1** have the same number-average molecular weight ($\overline{M}_n = 14$ kg mol⁻¹), the overall shape of the UV-Vis absorption spectra are also different. A substantial shift of the absorption maximum (up to 8 nm) was observed for **P2H-L1** relative to **P2H-L_{REF}**. This behaviour indicates improved regioregularity of **P2H-L1**. Furthermore, **L3**, **L4** and **L5** led to better molecular weights and regioregularity of the BDT-TID copolymer (Table S4;† Fig. 4). **P2H-L3**, **P2H-L4** and **P2H-L5** show a small bathochromic shift of the maximum of absorption (up to 2 nm) when compared to **P2S**. As with the **P1H** series, since the molecular weight of each entry is comparable, one can assume that the slight

	P type	N type
	Average ($\pm$ standard deviation) hole mobility ^a , cm ² V ⁻¹ s ⁻¹	Average ($\pm$ standard deviation) electron mobility ^a , cm ² V ⁻¹ s ⁻¹
P1S	$3.8 (\pm 0.8) \times 10^{-2}$	$4.0 (\pm 0.2) \times 10^{-3}$
P1H-L_{REF}	$4.3 (\pm 0.1) \times 10^{-3}$	$1.9 (\pm 0.1) \times 10^{-3}$
P1H-L3	$1.5 (\pm 0.9) \times 10^{-2}$	$1.0 (\pm 0.4) \times 10^{-3}$
P1H-L4	$9.0 (\pm 0.2) \times 10^{-3}$	$8.3 (\pm 0.3) \times 10^{-3}$
P2S	$3.1 (\pm 0.7) \times 10^{-2}$	$2.0 (\pm 0.1) \times 10^{-3}$
P2H-L_{REF}	$1.2 (\pm 0.1) \times 10^{-2}$	$1.5 (\pm 0.2) \times 10^{-3}$
P2H-L3	$1.3 (\pm 0.3) \times 10^{-2}$	$3.7 (\pm 0.3) \times 10^{-4}$
P2H-L4	$2.5 (\pm 0.2) \times 10^{-2}$	$1.4 (\pm 0.1) \times 10^{-3}$
P2H-L5	$2.0 (\pm 0.3) \times 10^{-2}$	$2.4 (\pm 0.5) \times 10^{-3}$
P3S (High $\overline{M}_n$)	$1.4 (\pm 0.2) \times 10^{-3}$	$2.9 (\pm 0.5) \times 10^{-4}$
P3S (Low $\overline{M}_n$)	$8.0 (\pm 0.9) \times 10^{-4}$	$9.4 (\pm 0.3) \times 10^{-5}$
P3H-L_{REF}	$1.2 (\pm 0.1) \times 10^{-4}$	$1.2 (\pm 0.1) \times 10^{-3}$
P3H-L4	$2.0 (\pm 0.4) \times 10^{-4}$	$4.3 (\pm 0.1) \times 10^{-5}$
P3H-L5	$6.2 (\pm 0.1) \times 10^{-5}$	$8.0 (\pm 0.1) \times 10^{-5}$
P4S-High	$2.5 (\pm 0.9) \times 10^{-4}$	$4.5 (\pm 1.1) \times 10^{-4}$
P4H-L2	$8.7 (\pm 1.0) \times 10^{-5}$	$2.3 (\pm 1.1) \times 10^{-4}$
P4H-L4	$7.5 (\pm 2.5) \times 10^{-5}$	$1.6 (\pm 0.6) \times 10^{-4}$
P4H-L5	$5.4 (\pm 0.7) \times 10^{-5}$	$5.6 (\pm 2.8) \times 10^{-4}$

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bathochromic shift observed for **P2H-L3** to **L5** to be the result of a more regioregular structure.

Fortunately, well-defined ^1H NMR spectra were obtained for this series. The ^1H NMR spectra of each monomer and polymer were obtained in deuterated 1,1,2,2-tetrachloroethane ($\text{C}_2\text{D}_2\text{Cl}_4$) at 90°C using the same acquisition parameters. In order to assign the signals and to identify defects (*i.e.* α - α homocouplings) homopolymers of BDT and TID were prepared as model compounds (see ESI†). Fig. 5 highlights the aromatic region of the ^1H NMR spectra of **P2S**, **P2H-L_{REF}**, **P2H-L2** and **P2H-L4**. For each polymer, the main peaks A, B and C correspond to the protons of the main chain while the residual signals (a–h) are attributed to end groups or α - α homocoupling units (see ESI†). These spectra show for **P2H-L2** and **P2H-L4**, a significant drop of the intensity of the peak at 8.21 ppm (a) (related TID–TID homocoupling) and of the shoulder near 7.7 ppm (c) (related to BDT–BDT homocoupling). For these two polymers, similar UV-Vis absorption spectra were obtained, indicating that the bulkier phosphines limit side reactions, particularly α - α

homocouplings. As with the **P1H** series, OFET measurements were used to evaluate the performance of the **P2H** series, with **P2S** being used as a benchmark (Table 1). With the exception of the lower electron mobility observed of **P2H-L3**, the overall performance of the **P2H** series is comparable to that of the reference polymer **P2S**. The overall OFET performance of the **P2H** series combined with the UV-Vis and ^1H NMR spectra tend to show once again that the bulkier phosphine leads to better materials with fewer defects when compared to **P2S**.

To push further our investigation, we then applied our catalytic system to the polymerization of Br_2 -BDT-OEH and DPP. As mentioned in the Introduction, the main objective was to evaluate if the electronic structure of the BDT side chain affects the reactivity and selectivity of the DHAP. The polymerization data are summarized in Table S5† and UV-Vis absorption spectra are shown in Fig. 6. Unlike **P1H** and **P2H** series, both $\overline{M}_n$ and polymerization yields (about 30%, soluble fraction recovered after Soxhlet extractions) of **P3H** series were lower than those obtained for **P3S** (Table S5†). For each entry, the

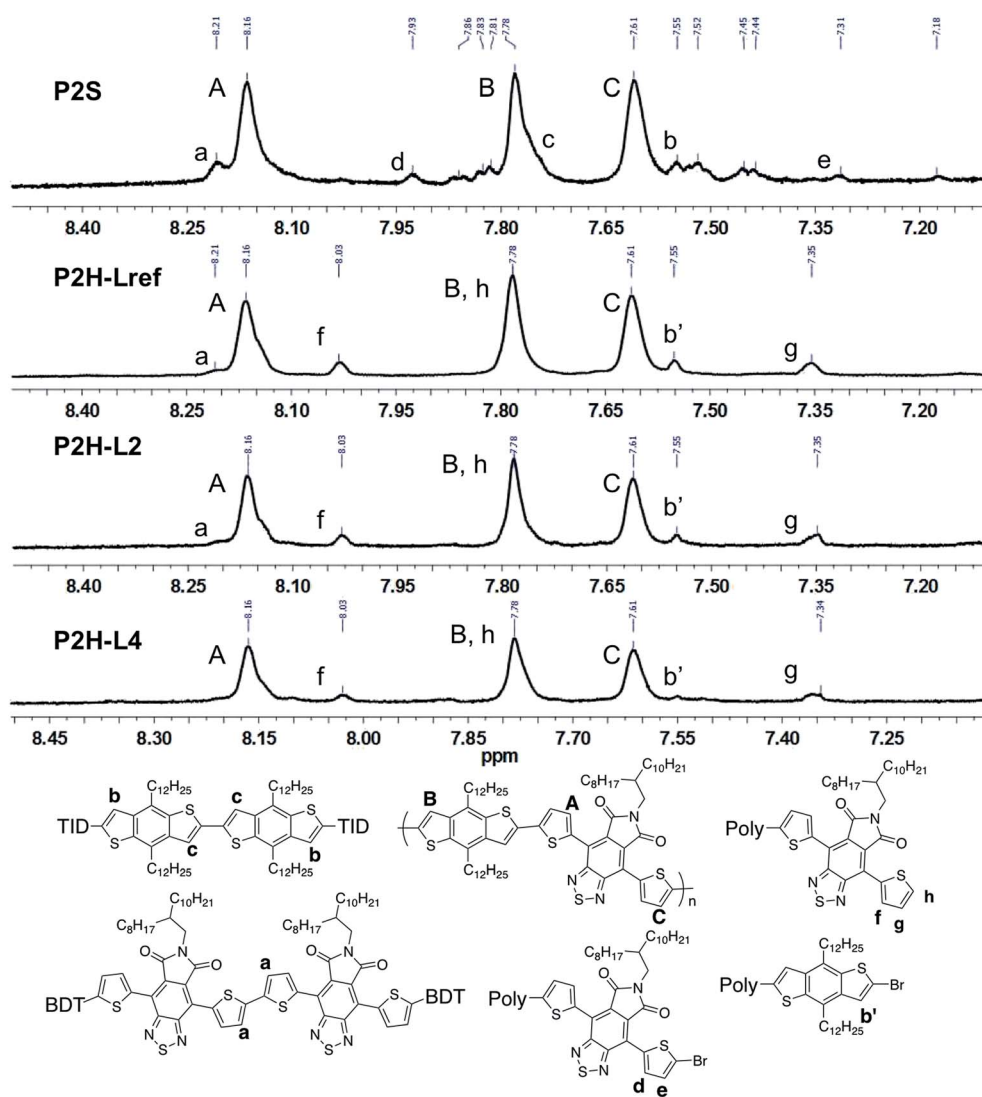


Fig. 5 ^1H NMR spectra of **P2S**, **P2H-L_{REF}**, **P2H-L2** and **P2H-L4**.



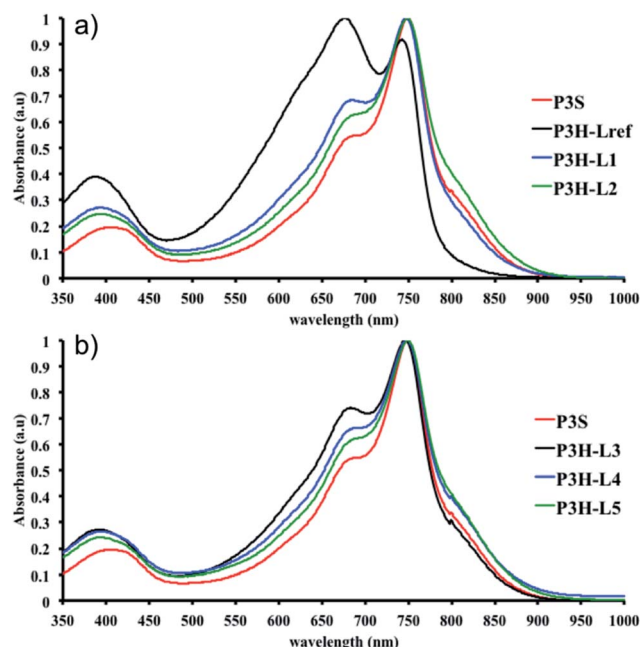


Fig. 6 UV-Vis spectra of P3S and P3H series (solution in CHCl_3).

polymerization was terminated after 2 hours due to gelation of the reaction mixture. Longer polymerization times did not lead to higher molecular weights. Moreover, as shown in Fig. 6, UV-Vis absorption spectra clearly demonstrate that the **P3H** series, regardless of the ligand, are not the same materials as **P3S** and the discrepancy between the UV-Vis spectra cannot be explained by the lower molecular weights alone. While no DPP-DPP homocoupling can be observed for **P3H-L_{REF}** (indicated by the absence of a shoulder at 820 nm), the shift of the absorption maximum (centered at 660 nm) indicates that the chemical structure of the conjugated skeleton differs from that of **P3S** and may be a result of lower DHAP selectivity between Br₂-BDT-OEH and DPP. Upon increasing the bulkiness of the alkoxy chain on the phosphine, the overall shape of the UV-Vis spectra approached that of **P3S** (Fig. 6). A maximum of absorption was found at 677 nm with a vibronic band at 740 nm for **P3H-L_{REF}**, while absorption maxima at 748 and 750 nm were observed for **P3HL1-L2** and **P3HL3-L5**, respectively, with a vibronic band at 685 nm. Based on these results, it seems that the alkoxy side chains on BDT modify the electronic properties of the monomer and affect the reactivity compared to **P1H** series. ¹H NMR spectra were fruitless in determining the nature of the defects present in the **P3S** and **P3H** series due to broad and featureless signals. As with **P1H** and **P2H**, OFET measurements were performed. **P3S** was used as a benchmark and compared to **P3H-L_{REF}**, **P3H-L4** and **P3H-L5** (Table 1). With the exception of the electron mobility of **P3H-L_{REF}** (the least well-defined polymer as determined from absorption spectra), the overall performance of the **P3H** series were one to two orders of magnitude lower than the reference polymer.

In order to verify that the electronic nature of the side chain installed on BDT moiety does affect the selectivity of the DHAP, we applied this catalytic system to the polymerization of Br₂-

BDT-OEH and TID (**P4H** series, Scheme 1). The polymerization data are summarized in Table S6[†] and UV-Vis absorption spectra are shown in Fig. 7. Molecular weights (from soluble fraction recovered after Soxhlet extraction) of the polymers obtained by the DHAP were lower than those obtained for **P4S**-High ($\overline{M}_n$ of 30 kg mol⁻¹) except for **P4H-L2** ($\overline{M}_n$ = 27 kg mol⁻¹) and **P4H-L4** ($\overline{M}_n$ = 30 kg mol⁻¹). Unlike **P4S**, the kinetics of DHAP was slow and no improvement of molecular weights was observed for longer polymerization times. Moreover, to eliminate the effect of molecular weight discrepancies during polymer comparison experiments, we also prepared **P4S**-Low, a low molecular weight batch ($\overline{M}_n$ = 16 kg mol⁻¹) similar to those obtained for **P4H-L1**, **L3** and **L5**. As shown in Fig. 7, the mismatch of all the UV-Vis absorption spectra of **P4H** series compared to **P4S**, combined with the shift of the intensity of the two main bands (400 nm and 640 nm) raises concerns as to the nature of the BDT side chain on the DHAP selectivity. **P4S**-High exhibits two maxima of absorption, one centered at 418 nm and another of higher intensity at 639 nm. These two maxima were observed at 414 nm and 617 nm for **P4S**-Low, indicating clear relationship between molecular weights and optical properties for **P4S**. As shown in Fig. 7a, the least comparable polymer was **P4H-L_{ref}**, with a number-average molecular weight of 10 kg mol⁻¹ and an UV-Vis absorption spectrum with two blue-shifted maxima (407 and 583 nm) compared to **P4S**-Low (414 and 617 nm). Another striking feature, aside from the hypsochromic shift, is the inversion of the intensity of the two maxima observed for **P4H-L_{ref}**. This clearly indicates that the chemical structure of the conjugated backbone of **P4H-L_{ref}** is not the same as **P4S** (High and Low). Moreover, despite similar number-average molecular weight when compared to **P4S**-High ($\overline{M}_n$ = 30 kg mol⁻¹), **P4H-L2** ($\overline{M}_n$ = 27 kg mol⁻¹) and **P4H-L4** ($\overline{M}_n$ = 30 kg mol⁻¹) also show hypsochromic shifts for both absorption

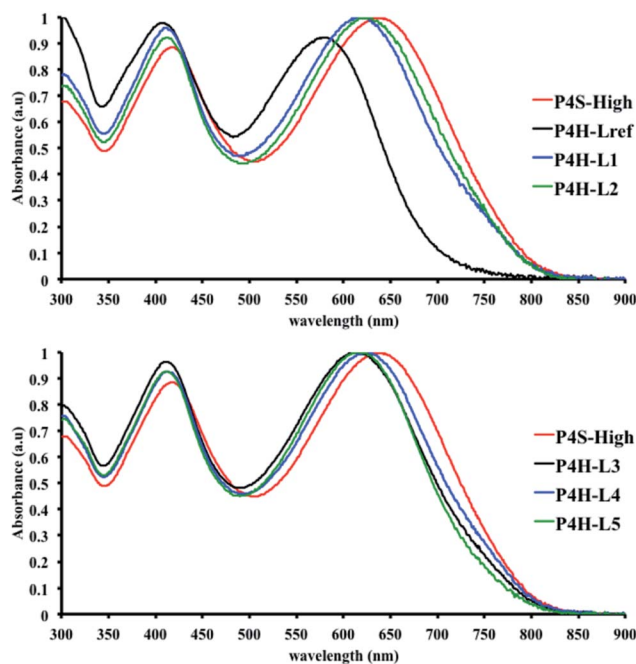


Fig. 7 UV-Vis spectra of P4S and P4H series (solution in CHCl_3).

bands, which also indicates the presence of defects within the main chain, as observed for the **P3H** series. Consistently with the fact that mismatches of the UV-Vis spectra between **P4S** and **P4H** series have been observed, lower OFETs performance for **P4H** series have been measured by experimental data (one order of magnitude lower than the reference polymer, see Table 1).

Well-defined ^1H NMR data were also obtained for **P4H**. Fig. 8 highlights the aromatic region of the ^1H NMR spectra of **P4S**, **P4H-L_{REF}**, **P4H-L2** and **P4H-L4**. On the basis of model compounds (see ESI†), the main peaks A, B and C correspond to the protons of the main chain while the residual signals (a–k) are related to end groups or homocoupling units. Although the exact integrations were difficult to determine, the lower intensity of the signal at 8.21 ppm for **P4H-L_{REF}** compared to the one

observed for **P4S** suggests that the DHAP may lead to lower amounts of TID–TID homocoupling side-reactions. Moreover, for **P4H-L2** and **P4H-L4**, a significant drop of the intensity of the peak at 8.21 ppm (a) (related to the homocoupling of TID) was observed compared to **P4S**. The other residual peaks (h: 7.77 ppm; f: 8.03 ppm; g: 7.35) for **P4H-L2** and **P4H-L4** are less intense compared to **P4H-L_{REF}** due to higher molecular weights.

To rationalize this experimental data, DFT calculations at the B3LYP/TZVP (DZVP for palladium) level have been carried out to evaluate the activation barrier of the C–H bond activation of the CMD pathway. This method has been previously successfully applied to the rationalization and prediction of the regioselectivity of direct (hetero)arylation reactions on different arenes and heteroarenes.⁶⁹ It is worth noting these calculations are

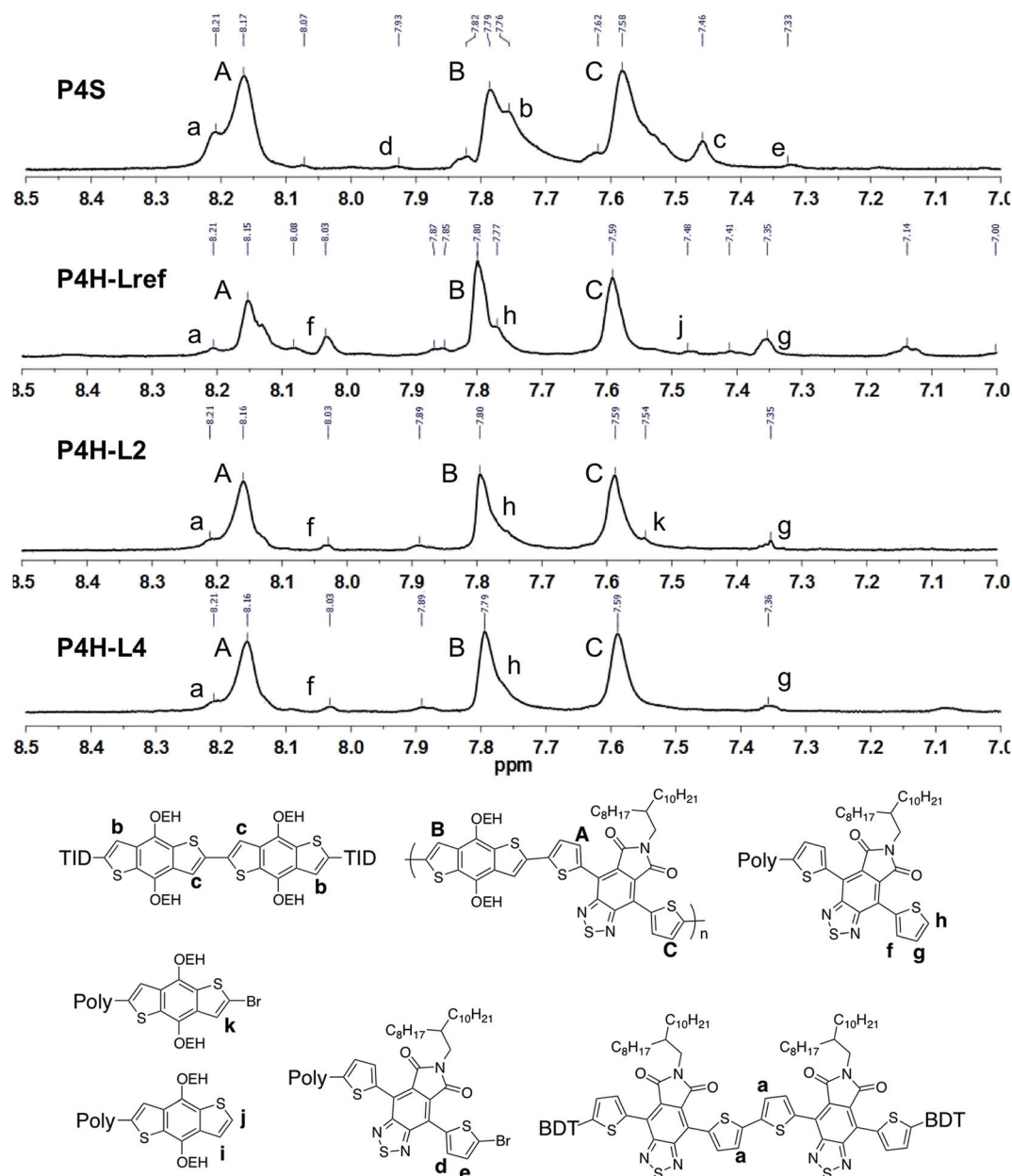


Fig. 8 ^1H NMR spectra of **P4S**, **P4H-L_{REF}**, **P4H-L2** and **P4H-L4**.



For **P4H** (Br₂-BDT-OEH with TID), the difference in the energy of activation (ΔE_a) between H_α and H_β for the TID moiety is 4.7 kcal mol⁻¹ (29.3–24.6 kcal mol⁻¹). For this system, a ratio of about 400 : 1 favoring H_α has been calculated. However, the ΔE_a between H_α of TID (24.6 kcal mol⁻¹) and the H_β of the Br₂-BDT-OEH (26.7 kcal mol⁻¹) is 2.1 kcal mol⁻¹ while the ΔE_a between H_α of TID (24.6 kcal mol⁻¹) and the H_β of the Br₂-BDT-C₁₂ (28.2 kcal mol⁻¹) is 3.6 kcal mol⁻¹. As discussed earlier, one can think that β-defects on BDT moiety can be the main defect found in the **P4H** series whereas a better selectivity is anticipated for the **P2H** series.

To continue our investigation, we carried out the polymerization of monomers with the bromide function on the electron-deficient units. 4,8-Bis(didodecyl)benzo[1,2-*b*:4,5-*b'*]dithiophene (BDT-C₁₂) or 4,8-bis(2-ethylhexyloxy)benzo[1,2-*b*:4,5-*b'*]dithiophene (BDT-OEH) was therefore polymerized with 3,6-bis(5-bromo-thiophen-2-yl)-2,5-bis(2-octyldodecyl)pyrrolo[3,4-*c*]-pyrrole-1,4-dione (Br₂-DPP) or 4,8-bis(5-bromo-thiophen-2-yl)-6-(2-octyldodecyl)-[1,2,5]thiadiazolo[3,4-*e*]isoindole-5,7-dione (Br₂-TID) to obtain the **P1H'**, **P2H'** and **P3H'** series (Scheme 1). The polymerization data for **P1H'** are reported in Table S7† and the UV-Vis absorption spectra in solution are shown in Fig. 10. The polymerization times were longer and the yields (about 25–30%) of the soluble fraction of the polymers in chloroform were consistently lower relative to both the **P1H** series and **P1S**, regardless of the nature of the ligand. DHAP polymerizations were run for 12 hours for all examples. After Soxhlet extractions, a large quantity of insoluble material was recovered indicating possible side reactions. Despite several attempts, both $\overline{M}_n$ and

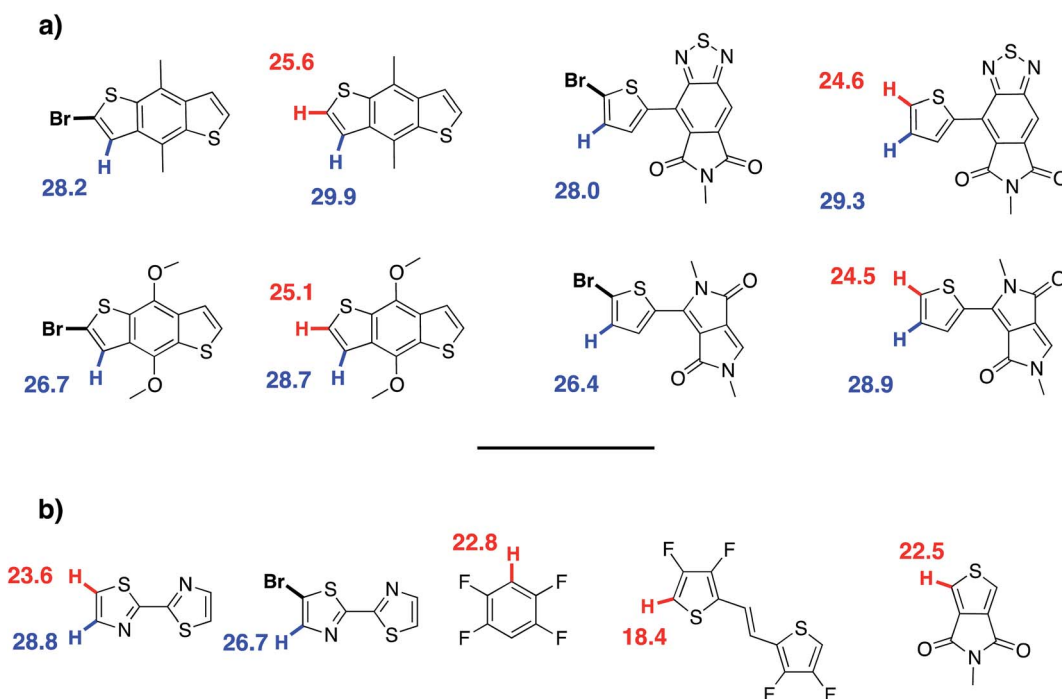


Fig. 9 The Gibbs free energy of the CMD transition state associated to the activation of α,β hydrogen atoms calculated by DFT.

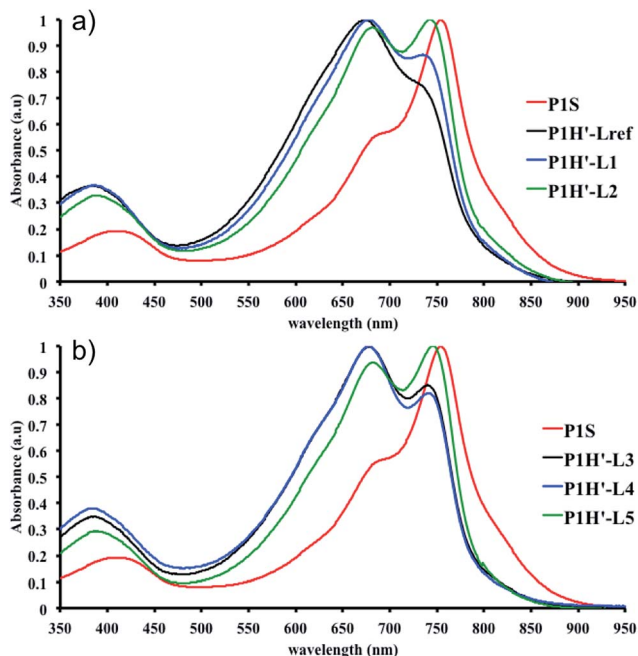


Fig. 10 UV-Vis spectra of **P1S** and **P1H'** series (solution in CHCl_3).

polymerization yield values were lower than those of the reference polymer **P1S**. As shown in Fig. 10, the UV-Vis absorption spectra of the **P1H'** series were different from the benchmark **P1S**. Clearly, this catalytic system did not yield the targeted polymers as opposed to the **P1H** series.

We then applied the catalytic system for the polymerization of 4,8-bis(didodecyl)benzo[1,2-*b*:4,5-*b'*]dithiophene (BDT-C₁₂) and 4,8-bis(5-bromo-thiophen-2-yl)-6-(2-octyldodecyl)-[1,2,5]thiadiazolo[3,4-*e*]isoindole-5,7-dione (Br₂-TID) **P2H'** (Scheme 1). The polymerization data for **P2H'** are reported in Table S8,[†] while the UV-Vis absorption spectra in solution are shown in Fig. 11. The polymerization times were generally longer and the yields (soluble fraction of the polymer in chloroform) lower compared to the **P2H** series and **P2S**. Like **P1H'** series, the UV-Vis absorption spectra obtained for **P2H'** were different from the reference, indicating, once again, that the targeted polymers were not obtained with this catalytic system.

As for the **P2H** series, well-defined ^1H NMR data were obtained for the **P2H'** samples in 1,1,2,2-tetrachloroethane ($\text{C}_2\text{D}_2\text{Cl}_4$) at 90 °C (see ESI, Section 4†). The shape of the UV-Vis absorption spectra of the **P2H'** series combined with ^1H NMR analyses confirm the presence of defects within the conjugated main chain. The same trend was observed for the copolymerization of BDT-OEH with $\text{Br}_2\text{-DPP}$ (**P3H'** series, see Scheme 1). Lower molecular weights and yields were obtained with all phosphines (Table S9†). As with **P1H'**, a large quantity of insoluble material was recovered after Soxhlet extractions and the UV-Vis absorption spectra of the soluble fraction did not match that obtained for **P3S** (Fig. 12). In the light of the result obtained for the **P1H'–P3H'** series, we did not synthesize the **P4H'** series, *i.e.* the equivalent polymers to **P4H** with the bromide moiety on the electron-poor unit.

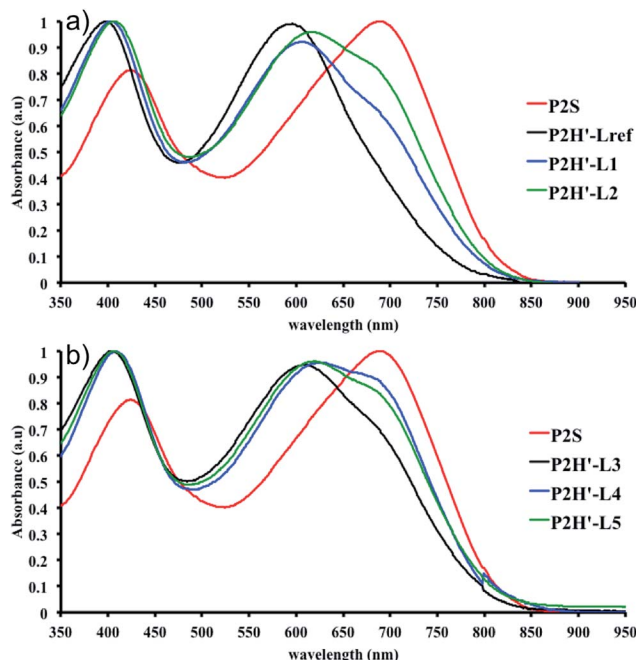


Fig. 11 UV-Vis spectra of P2S and P2H' series (solution in CHCl_3).

DFT calculations performed for **P1H'** (BDT-C₁₂ with Br₂-DPP) helped again to rationalize the experimental data obtained for this series (Fig. 9). Here, the ΔE_a between the H_α (25.6 kcal mol⁻¹) and H_β (29.9 kcal mol⁻¹) of BDT-C₁₂ is 4.3 kcal mol⁻¹, indicating once again an excellent selectivity for the activation of H_α. However, the ΔE_a between H_β of the Br₂-DPP (26.4 kcal mol⁻¹) and H_α of the BDT-C₁₂ (25.6 kcal mol⁻¹) is only of 0.8 kcal mol⁻¹, suggesting poor selectivity and significant unwanted DPP-DPP β-defects.

For **P2H'** series (BDT-C₁₂ with Br₂-TID), the ΔE_a between H_α (25.6 kcal mol⁻¹) and H_β (29.9 kcal mol⁻¹) of BDT-C₁₂ is 4.3 kcal mol⁻¹. On the other hand, the ΔE_a between H_β of the Br₂-TID (28.0 kcal mol⁻¹) and H_α of the BDT-C₁₂ (25.6 kcal mol⁻¹) is 2.4 kcal mol⁻¹, leading to a 20 : 1 selectivity at 125 °C which would induce some TID-TID β-defects. The same trend was observed for the copolymerization between BDT-OEH with Br₂-DPP (**P3H'** series, see Scheme 1). Although the ΔE_a between H_α (25.1 kcal mol⁻¹) and H_β (28.7 kcal mol⁻¹) of BDT-OEH of 3.6 kcal mol⁻¹ leads to a selectivity of about 100 : 1, the ΔE_a between H_β of the Br₂-DPP (26.4 kcal mol⁻¹) and H_α of the BDT-OEH (25.1 kcal mol⁻¹) of 1.3 kcal mol⁻¹ gives a poor 5 : 1 selectivity, which would induce possible DPP-DPP β-defects. These observations are in good agreement with the experimental data obtained for **P3H'** series. Finally, according to the theoretical calculations, one can think that the synthesis of **P4H'** by DHAP (BDT-OEH with Br₂-TID) would also suffer from selectivity issues. Indeed, the ΔE_a between H_β of the Br₂-TID (28.0 kcal mol⁻¹) and H_α of the BDT-OEH (25.1 kcal mol⁻¹) of 2.9 kcal mol⁻¹ gives a 40 : 1 selectivity, indicating possible TID-TID β-defects.

On the basis of these experimental-theoretical data, we extended the scope of our DFT calculations to other BDT and DPP-based copolymers recently synthesized by DHAP in order to

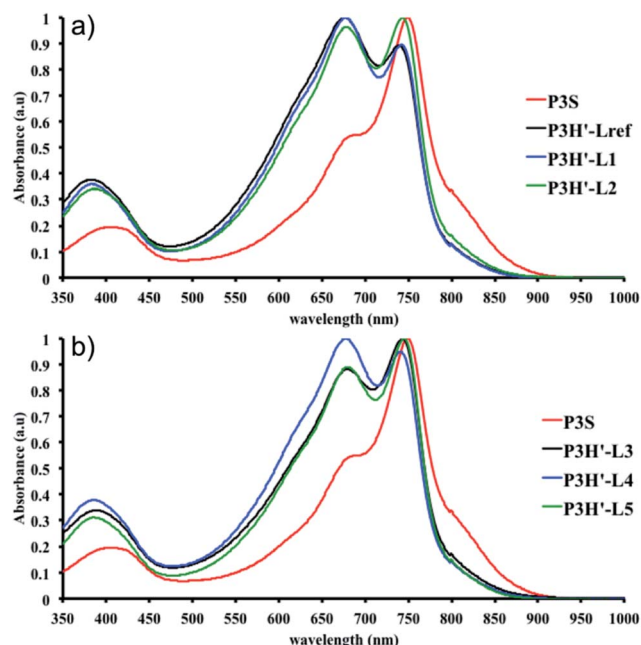
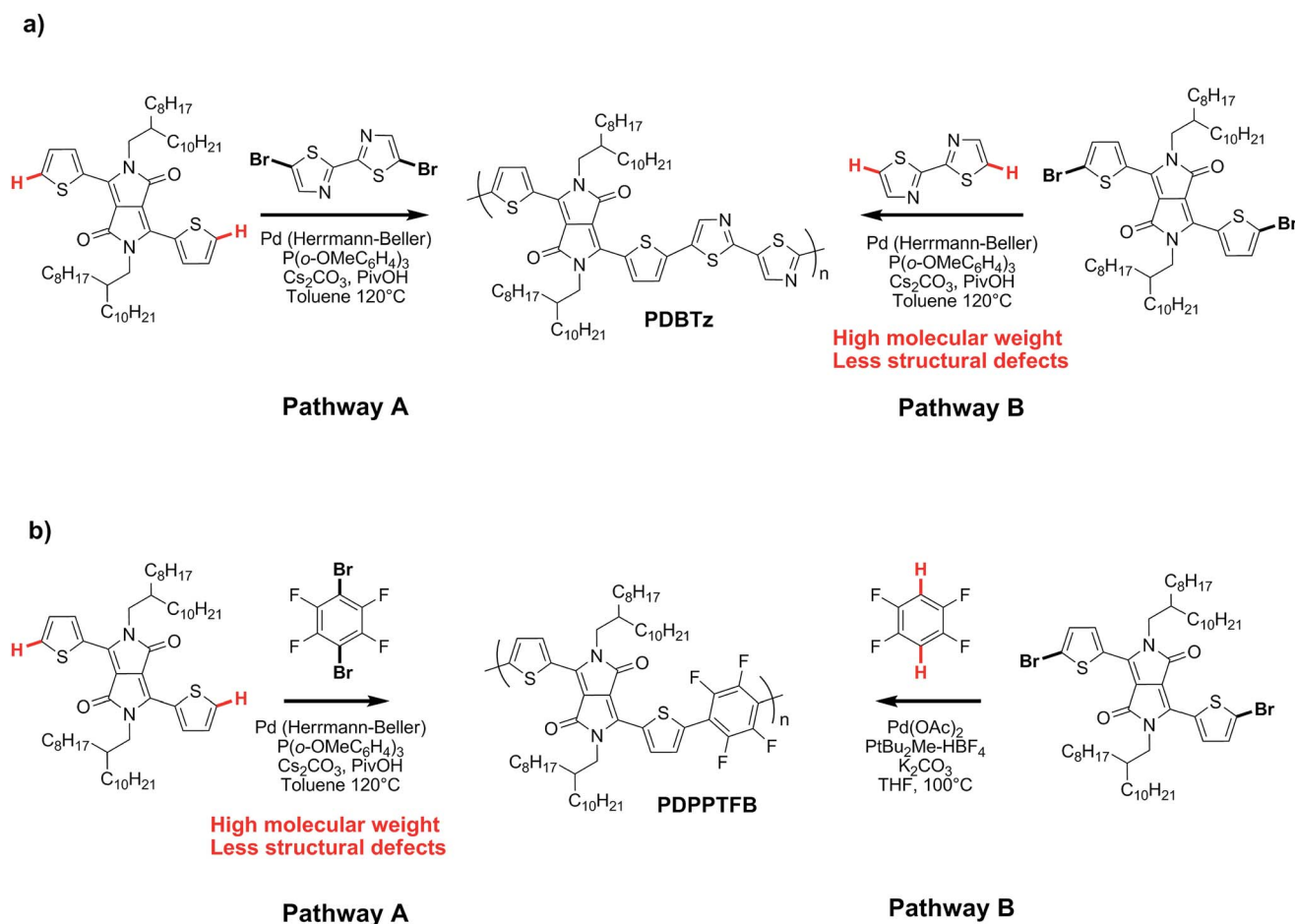


Fig. 12 UV-Vis spectra of P3S and P3H' series (solution in CHCl_3).

explain why some of them can be successfully prepared while other cannot. For instance, Li *et al.* reported that a relatively well-defined copolymer was obtained from the copolymerization of Br_2 -DPP with bithiazole while poor performance was reported for the same copolymer synthesized using Br_2 -bithiazole and DPP (Scheme 2a).⁷⁰ For the first case, the ΔE_a between the H_α (23.6 kcal mol⁻¹) and H_β (28.8 kcal mol⁻¹) of bithiazole moiety is 5.2 kcal mol⁻¹ which means that the activation of the H_α position is highly selective. Meanwhile, a selectivity of about 35 : 1 is observed when considering the ΔE_a value of 2.8 kcal mol⁻¹ between H_β of the Br_2 -DPP (26.4 kcal mol⁻¹) and H_α of the bithiazole (23.6 kcal mol⁻¹). On the other hand, for the Br_2 -bithiazole/DPP system, the ΔE_a between H_β of the Br_2 -bithiazole (26.7 kcal mol⁻¹) and the H_α of the DPP (24.5 kcal mol⁻¹) is only of 2.2 kcal mol⁻¹ and should lead to less-defined structure. Wang and co-workers have also reported that the synthetic pathways used for the synthesis of DPP-tetrafluorobenzene (TFB) copolymer by DHAP affect the regioregularity of the resulting copolymer (see Scheme 2b) and, once again, the experimental data are in good agreement with the DFT calculations.⁷¹ Indeed, for pathway A (DPP and Br_2 -TFB), the ΔE_a between H_α and H_β for the DPP moiety is 4.4 kcal mol⁻¹ (24.5 vs. 28.9 kcal mol⁻¹, respectively) and leads to a selectivity ratio of



Scheme 2 Synthetic routes for (a) PDBTz and (b) PDPPTFB by DHAP.^{70,71}



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