

INORGANIC CHEMISTRY

FRONTIERS



We herein report a new Cu^{2+} MOF $\{[\text{Cu}_3(\text{PEIP})_2(5\text{-NH}_2\text{-mBDC})(\text{DMF})]\cdot 7\text{DMF}\}_\infty$ denoted as **Cu-PEIP**·7DMF which represents the initial 3d metal complex with the ligand PEIPH₂. Although this compound was prepared by a serendipitous self-assembly synthetic procedure, after its structure was known the synthesis of an analogous complex was targeted with high priority and achieved following a rational synthetic method. **Cu-PEIP** exhibits a 3D-framework with a unique trinodal underlying network and point symbol $(4\cdot 5^2)_4(4\cdot 5^2\cdot 6^4\cdot 8^3\cdot 9^2)_2(5^2\cdot 8^4)$. This network consists of pillared **kgm-a** layers and thus resembles the structure of HKUST-1. **Cu-PEIP** shows a significant BET area ($1785\text{ m}^2\text{ g}^{-1}$) as well as high CO₂ sorption capacity (4.75 mmol g^{-1} at 273 K) and CO₂/CH₄ selectivity (8.5 at zero coverage and 273 K).

Materials

Low-pressure argon, hydrogen, carbon dioxide and methane adsorption measurements were carried out on an Autosorb 1-MP instrument from Quantachrome equipped with multiple pressure transducers for highly accurate analyses and an oil-free vacuum system. High pressure excess adsorption measurements were conducted manometrically on a PCTPro-2000 instrument (Setaram) with the aid of a regular valve sealed stainless-steel measuring cell. The NIST database was implemented for estimating the compressibility of gases (H_2 , CH_4 , CO_2) and helium was used for dead volume calibrations at 273–303 K. In order to avoid potential helium sorption errors, the dead volume at 77 K was calculated through a reference curve obtained by using different volumes

of non-adsorbing materials (Pyrex glass). Ultra-high purity grade Ar (99.999%), He (99.999%), H₂ (99.999%), CO₂ (99.999%) and CH₄ (99.9995%) were used for all adsorption measurements. Prior to analysis, as-made **Cu-PEIP** was soaked in CH₂Cl₂ at room temperature for three (3) days during which the supernatant solution was replaced six (6) times. The dichloromethane suspended samples were transferred inside the chamber of a supercritical CO₂ dryer (Bal-Tec CPD 030) and CH₂Cl₂ was exchanged with liquid CO₂ over a period of 5 hours at 8 °C. During this period, liquid CO₂ was vented under positive pressure every 5 minutes. The rate of CO₂ venting was always kept below the rate of filling so as to maintain full drying conditions inside the chamber. Following venting, the temperature was raised to 40 °C (above the critical temperature of CO₂), kept there for 1 hour and then slowly vented over the period of 1 hour. The dried sample was transferred immediately inside a pre-weighted, argon filled 9 mm cell and closed using CellSeal™ provided by Quantachrome to prevent the intrusion of oxygen and atmospheric moisture during transfers and weighing. The cell was then transferred to the outgassing station where the sample was evacuated under dynamic vacuum at room temperature until the outgas rate was less than 2 mTorr min⁻¹. After evacuation, the sample and cell were re-weighed to obtain the precise mass of the evacuated sample. Finally, the tube was transferred to the analysis port of the gas adsorption instrument. Likewise, for high pressure measurements, the dried sample was transferred sealed to an argon filled glove box (Labstar, MBraun, H₂O and O₂ <0.5 ppm), and a quantity of approx. 180 mg was inserted into a pre-weighted stainless-steel container. The sample mass was calculated and the container was sealed with a high pressure valve. The whole cell was removed from the glove box and was attached to the high pressure manometric apparatus. All-metal face seals were used. After suitable evacuation of the apparatus the sample was further outgassed overnight at room temperature under high vacuum ($p < 10^{-6}$ mbar). The outgassing procedure was repeated after each high pressure measurement.

Single crystal X-ray crystallography

Single crystal X-ray diffraction data were collected on an Oxford-Diffraction Supernova diffractometer, equipped with a CCD area detector utilizing Cu K α ($\lambda = 1.5418$ Å) radiation. A suitable crystal was mounted on a Hampton cryoloop with Paratone-N oil and transferred to a goniostat where it was cooled for data collection. Empirical absorption corrections (multiscan based on symmetry-related measurements) were applied using CrysAlis RED software.¹⁸ The structure was solved by direct methods using SIR2004¹⁹ and refined on F^2 using full-matrix least-squares with SHELXL97.²⁰ Software packages used were as follows: CrysAlis CCD for data collection,¹⁸ CrysAlis RED for cell refinement and data reduction,¹⁸ WINGX for geometric calculations,²¹ and DIAMOND²² and X-Seal²³ for molecular graphics. The non-H atoms were treated anisotropically, whereas the aromatic H atoms were placed in calculated, ideal positions and refined as riding on

Table 1 Selected crystal data for **Cu-PEIP**

Complex	Cu-PEIP
Empirical formula	C ₃₉ H ₂₅ Cu ₃ N ₆ O ₁₃
Formula weight	976.30
Temperature (K)	100(2)
Radiation	Cu K α ($\lambda = 1.54180$ Å)
Crystal system	Monoclinic
Space group	$I2/m$
a (Å)	13.657(5)
b (Å)	18.686(5)
c (Å)	31.503(5)
β (°)	101.532(5)
V (Å ³)	7877(4)
Z	4
D_c (g cm ⁻³)	0.823
μ (mm ⁻¹)	1.266
Refls coll.	27 001
Unique refls	7255
R_{int}	0.0367
R_1^a [$I > 2\sigma(I)$]	0.0749
wR_2^b (all data)	0.2441
GOF	1.157
$\Delta\rho_{min/max}$ (e Å ³)	1.279/−0.658

$$^a R_1 = \sum ||F_o| - |F_c|| / \sum |F_o|, \quad ^b wR_2 = \{ \sum [w(|F_o|^2 - |F_c|^2)^2] / \sum [w(|F_o|^4)] \}^{1/2}.$$

their respective carbon atoms. Electron density contributions from disordered guest molecules were handled using the SQUEEZE procedure from the PLATON software suit.²⁴ Selected crystal data for **Cu-PEIP** are summarized in Table 1. CCDC 1492970 contains the supplementary crystallographic data for this paper.

Results and discussion

Over the last few years, we have been systematically investigating the use of semi-rigid polytopic ligands, such as CIPH₃ as a method for the synthesis of new MOFs.^{9c,17} An extension of these investigations included the use of PEIPH₂, which is similar to CIPH₃ but contains a pyridyl group in the place of the benzoic acid moiety, in MOF chemistry. Compound **Cu-PEIP** was initially prepared from the reaction of Cu(NO₃)₂·2.5H₂O and PEIPH₂ in DMF at 100 °C in 70% yield. A similar reaction was performed that contained instead of the pre-formed Schiff base ligand PEIPH₂ its constituent moieties, *i.e.* 5-NH₂-mBDCH₂ and 4-pyridinecarboxaldehyde aiming at the *in situ* formation of the ligand and the isolation of the final product through a one-pot reaction (by omitting the ligand preparation step). This was realized from the reaction of Cu(NO₃)₂·2.5H₂O, 5-NH₂-mBDCH₂ and 4-pyridinecarboxaldehyde in DMF at 100 °C in 65% yield. Interestingly, **Cu-PEIP** was isolated as excellent quality green polyhedral-like crystals in very high yields with both synthetic methods. When the structure of the compound **Cu-PEIP** was known the synthesis of an analogous complex was targeted and achieved by following a rational synthetic procedure. This procedure involved the use in the reaction mixture of the reduced analogue of PEIPH₂, *i.e.* ligand PIPH₂ that remains intact in solution, and 5-NH₂-mBDCH₂. Thus, compound **Cu-PIP** was prepared from the



reaction of $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$, PIPH_2 and $5\text{-NH}_2\text{-mBDC}$ in DMF at 100°C in 46% yield. The identity and purity of the bulk products of **Cu-PEIP** and **Cu-PIP** and their structural relation were confirmed by PXRD, elemental analysis and infrared spectroscopy (Fig. S1–S4 in the ESI†).

Compound **Cu-PEIP** crystallizes in the monoclinic space group $I2/m$. There are two crystallographically unique Cu^{2+} ions in the structure (Cu1 and Cu2), both adopting a square pyramidal coordination geometry (Fig. 1): Cu1 is coordinated with four carboxylic oxygen atoms $2 \times (\text{O1-O2})$ of four different PEIP^{2-} ligands and one terminal DMF (O7) solvent molecule, whereas Cu2 is connected with two carboxylate oxygen atoms (O3-O4) from two different PEIP^{2-} anions, two carboxylate oxygen atoms (O5-O6) from two different $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands and one nitrogen atom from a third PEIP^{2-} ligand (N1). This connectivity gives rise to a 3-D framework, which comprises two paddle-wheel $[\text{Cu}_2(\text{COO})_4]$ SBUs and exhibits large channels running mainly along the a -axis (Fig. 2a). The first type of dinuclear SBU, $[\text{Cu}_2]_A$, is formed by two symmetry-equivalent Cu1 metal ions, which are bridged by four *syn, syn*- $\mu_2\text{-COO}^-$ groups originating from four PEIP^{2-} ligands, whereas the second paddle-wheel type SBU, $[\text{Cu}_2]_B$, consists of two symmetry-equivalent Cu2 ions, which are bridged by four *syn, syn*- $\mu_2\text{-COO}^-$ groups originating from two PEIP^{2-} and two $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands. The axial positions in $[\text{Cu}_2]_A$ are occupied by two oxygen atoms from two DMF molecules, while those in $[\text{Cu}_2]_B$ by two pyridyl nitrogen atoms from two PEIP^{2-} ligands. The $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands bridge two $[\text{Cu}_2]_B$ SBUs with a $[\text{Cu}_2]_B \cdots [\text{Cu}_2]_B$ separation of $\sim 9.3 \text{ \AA}$ while the isophthalate

moiety of the PEIP^{2-} ligands bridges a $[\text{Cu}_2]_A$ to a $[\text{Cu}_2]_B$ SBU with the $[\text{Cu}_2]_A \cdots [\text{Cu}_2]_B$ separation being $\sim 9.2 \text{ \AA}$. In this arrangement, the paddle-wheels and the isophthalate moieties of both ligands create an undulated (3·6·3·6) semi-regular plane net (**kgm-a** lattice) parallel to the bc plane (Fig. 2b). This layer contains two types of cavities; a hexagonal shaped with a relatively large diameter of $\sim 8\text{--}9 \text{ \AA}$ surrounded by six trigonal shaped ones with a smaller diameter of $\sim 4\text{--}5 \text{ \AA}$ as found by PLATON²⁴ (taking into account the van der Waals radii of the atoms). The trigonal shaped cavities are composed of one $[\text{Cu}_2]_A$ and two $[\text{Cu}_2]_B$ SBUs and are filled with the terminal DMF molecules bound to $[\text{Cu}_2]_A$ SBUs. The hexagonal shaped cavities are composed of two $[\text{Cu}_2]_A$ separating two pairs of two $[\text{Cu}_2]_B$ SBUs. The sides of the hexagonal cavities comprise the isophthalate moieties of two approximately parallel $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands separating two adjacent isophthalate moieties of two pairs of PEIP^{2-} ligands.

The **kgm-a** layers are pillared through the pyridyl groups of the PEIP^{2-} ligands giving rise to hexagonal channels running parallel to the a axis (Fig. 2c and d). Since there are only four PEIP^{2-} ligands composing the hexagonal cavities, only four $[\text{Cu}_2]_B$ SBUs act as connection points in the pillaring. The $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands point their amino group toward the hexagonal cavities, thus compartmentalizing the cavities and the hexagonal channels as shown in Fig. 2c and d. Since part of the isophthalate ligands and part of the $[\text{Cu}_2]$ SBUs participate in the pillaring there is enough space left uncovered giving rise to pores through the other two directions of the material. The pores accommodate DMF molecules.

The solvent-accessible volume calculated by PLATON²⁴ is 5241 \AA^3 and corresponds to 66.5% of the unit-cell volume (7877.0 \AA^3). A representation of the voids using the structure visualization program MERCURY²⁵ reveals a continuous and complex 3D pore network, where the large compartmentalized hexagonal cavities communicate through relatively wide channels with a diameter $\geq 4 \text{ \AA}$ (Fig. 3).

From the topological point of view the $[\text{Cu}_2]_A$ paddle-wheel SBUs possessing two DMF molecules on the Cu1 apical sites serve as 4-coordinated nodes, while the $[\text{Cu}_2]_B$ SBUs possessing two pyridyl N atoms on the apical positions of the Cu2 ions act as 6-c nodes. The PEIP^{2-} ligands bridge a $[\text{Cu}_2]_A$ to a $[\text{Cu}_2]_B$ through the isophthalate moiety and a $[\text{Cu}_2]_B$ through the pyridyl N atom, thus serving as a 3-c node. The $5\text{-NH}_2\text{-mBDC}^{2-}$ ligand simply bridges two $[\text{Cu}_2]_B$ SBUs and therefore is topologically silent (serves as a bridge). In this arrangement a 3,4,6-coordinated trinodal network with the stoichiometry $(3\text{-c})_4(4\text{-c})(6\text{-c})_2$ and point symbol $(4\cdot5^2)_4(4^2\cdot5^4\cdot6^4\cdot8^3\cdot9^2)_2(5^2\cdot8^4)$ forms which is, so far, unique (Fig. S5–S8 in the ESI†). The network adopted by **Cu-PEIP** resembles the **tbo** net adopted by HKUST-1 and the **eea** net.^{12d,e,15,16} All three nets can be considered as pillared **kgm-a** layers. The augmented Kagomé lattice (**kgm-a**) is based on $[\text{Cu}_2(\text{COO})_4]$ SBUs and isophthalate moieties which alternate to create big hexagonal cavities comprising six paddle-wheels and six isophthalate moieties. In **tbo** the **kgm-a** layers are separated (pillared) by the third carboxylate of the trimesic acid while in the **eea** net the layers are

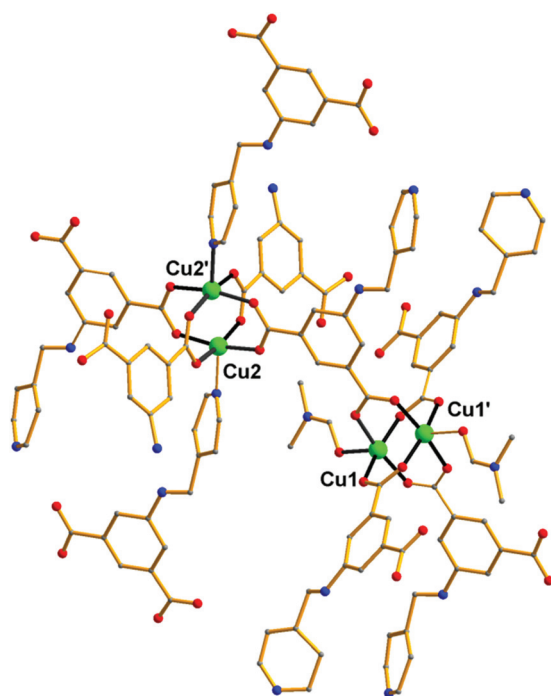


Fig. 1 Representation of the connectivity of the $[\text{Cu}^{2+}]_2$ paddle wheel SBUs through the PEIP^{2-} ligand in **Cu-PEIP**. Colour code: Cu green, O red, N blue, C gray. H atoms are omitted for clarity.



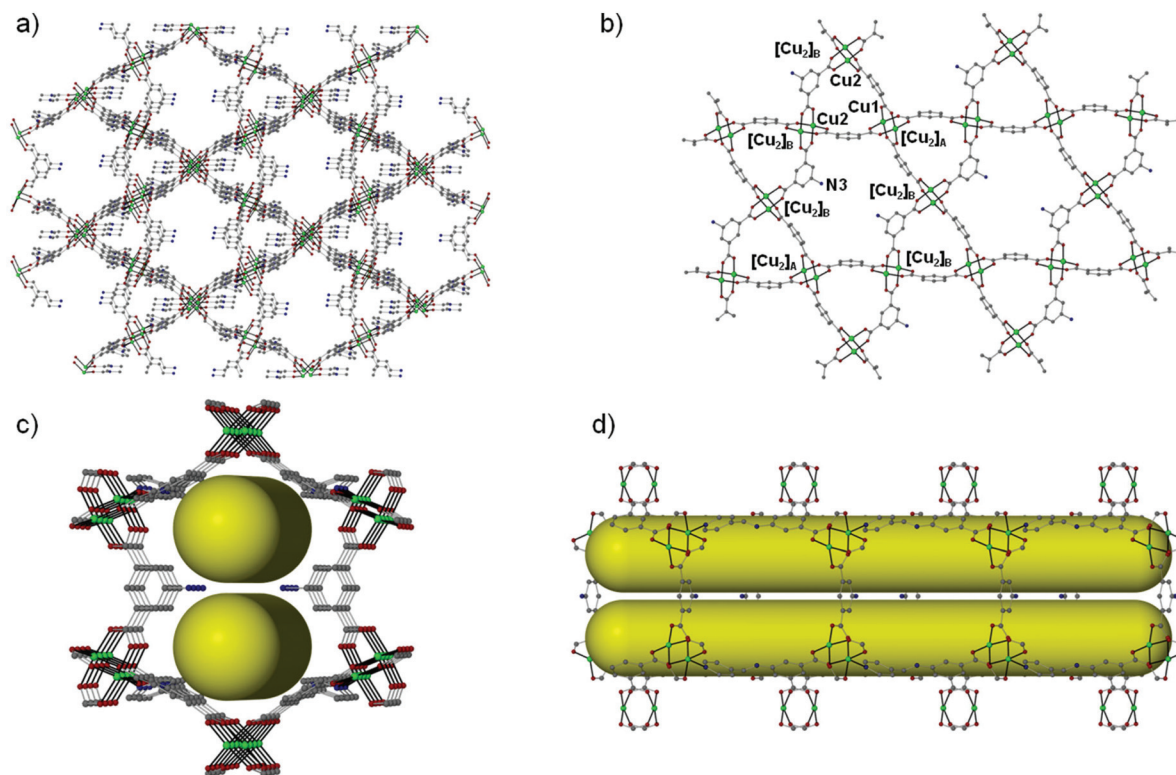


Fig. 2 Representations of (a) the 3D framework along the *a* axis, (b) the (3-6-3-6) semi-regular net (augmented Kagomé lattice: **kgm-a**), (c) the compartmentalized hexagonal channels along the *a* axis and (d) the compartmentalized channel projected along the *a* axis for complex **Cu-PEIP**. Colour code: Cu green, O red, N blue, C gray. H atoms are omitted for clarity.

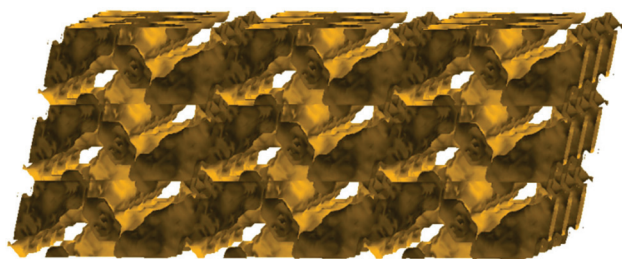


Fig. 3 Representation of the pore network (shown in yellow) of **Cu-PEIP** (only pores and channels with diameter ≥ 4 Å are shown) along the *b* axis.

separated by the pending pyridyl groups of the pyridyl-isophthalate ligands. In both **tbo** and **eea** nets, all six bridging ligands around each hexagonal cavity serve as pillars to both sides of the layer while in **Cu-PEIP** only four PEIP^{2-} ligands serve as pillars since two out of the six bridges around the hexagonal cavities are $5\text{-NH}_2\text{-mBDC}^{2-}$ ligands which cannot connect neighboring **kgm-a** layers.

The thermal stability of **Cu-PEIP** and **Cu-PIP** was investigated by means of the thermogravimetric analysis (TGA) technique (Fig. S9 and S10 in the ESI†). The TGA curves for the two complexes are essentially identical and thus only that of **Cu-PEIP** shall be discussed in detail. The TGA curve of **Cu-PEIP** indicates that this compound is decomposed through a multi-

step process which is completed at a fairly low temperature (~ 400 °C). The first two steps that are related to the removal of the lattice and bound DMF solvent molecules appear in the temperature range 30 °C to 260 °C and correspond to $\sim 40\%$ of the material's total mass. This value is in agreement with the corresponding calculated value for **Cu-PEIP-7DMF** ($\sim 39.2\%$). The next steps are associated with the decomposition of the two bridging ligands PEIP^{2-} and $5\text{-NH}_2\text{-mBDC}^{2-}$ and are completed at ~ 400 °C.

The relatively large solvent accessible volume present in **Cu-PEIP** prompted us to investigate its gas sorption properties. Several activation methods were employed, however the most efficient one involved the use of dichloromethane to remove the guest solvent molecules followed by treatment with supercritical CO_2 . Argon sorption measurements at 87 K revealed a type-I isotherm (Fig. 4), typical for a microporous solid, from which the apparent BET area was found to be $1785 \text{ m}^2 \text{ g}^{-1}$ (Langmuir, $1814 \text{ m}^2 \text{ g}^{-1}$), close to the geometric surface area ($2059 \text{ m}^2 \text{ g}^{-1}$) calculated from the single crystal structure using Poreblazer.²⁶ The total pore volume is $0.64 \text{ cm}^3 \text{ g}^{-1}$ at relative pressure, p/p_0 , 0.99, which is slightly lower compared to the value of $0.75 \text{ cm}^3 \text{ g}^{-1}$ calculated²⁶ from the crystal structure. Given that the PXRD pattern of the activated sample suggests an intact framework, the difference in the pore volume could be explained by trapped organic molecules inside the pores of **Cu-PEIP**. The pore size distribution, calculated using Non-



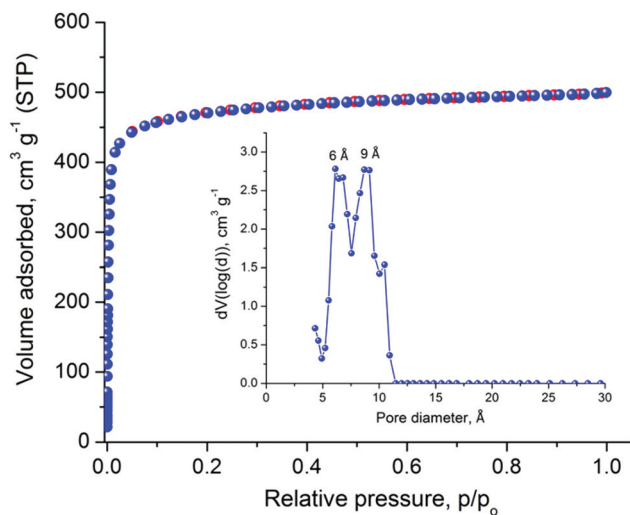


Fig. 4 Argon sorption isotherm of Cu-PEIP recorded at 87 K and pore size distribution curve calculated by NLDFT (inset).

Local Density Functional Theory (NLDFT) after a successful fitting of the Ar adsorption isotherm data using a suitable NLDFT kernel (Fig. S11 in the ESI†), shows two major peaks centered at 6 Å and 9 Å (inset in Fig. 4), in agreement with the crystallographic analysis.

The high porosity of Cu-PEIP in combination with the presence of unsaturated Cu(II) sites and -NH_2 groups, prompted us to further investigate the gas sorption properties by recording low and high pressure isotherms of CO_2 , CH_4 and H_2 at different temperatures from which the total uptake, isosteric heat of adsorption (Q_{st}) and CO_2/CH_4 selectivity were calculated. We note that amine functionalized MOFs are highly desirable, especially for CO_2 capture, because high uptake and selectivity are expected due to favorable acid–base interactions.²⁷

The CO_2 uptake at 1 bar is 4.75 mmol g^{-1} (20.9 wt%) and 2.80 mmol g^{-1} (12.3 wt%) at 273 K and 298 K, respectively (Fig. 5). These values are within the same range of representative and high performance NH_2 -functionalized MOFs including $\text{NH}_2\text{-MIL-101}(\text{Cr})$ (3.2 and 1.9 mmol g^{-1} at 273 K and 298 K),²⁸ $\text{Zn}(\text{Atz})_2$ (4.35 mmol g^{-1} at 1.2 bar and 273 K),²⁹ $\text{NH}_2\text{-MIL-125}(\text{Ti})$ (5.9 mmol g^{-1} at 273 K)³⁰ and $\text{NH}_2\text{-UiO-66}$ (between 2.89 and 3.04 mmol g^{-1} at 298 K).³¹ The isosteric heat of adsorption, Q_{st} , was calculated to be 34.2 kJ mol^{-1} at zero coverage (Fig. 6 and S12). Such a moderate value, which is lower compared to some NH_2 -functionalized MOFs, such as $\text{NH}_2\text{-MIL-101}(\text{Cr})$ (52 kJ mol^{-1})²⁸ and $\text{Zn}(\text{Atz})_2$ (40.8 kJ mol^{-1}),²⁹ is highly desirable because of the anticipated lower regeneration energy demand.²⁷ At elevated pressures, very high gravimetric and volumetric CO_2 uptake is observed, reaching $317 \text{ cm}^3 \text{ g}^{-1}$ (14.2 mmol g^{-1}) and $242 \text{ cm}^3 \text{ cm}^{-3}$, respectively, at 298 K and 25 bar, while the corresponding saturation uptake at 50 bar is $347 \text{ cm}^3 \text{ g}^{-1}$ (15.5 mmol g^{-1}) and $264 \text{ cm}^3 \text{ g}^{-1}$ (Fig. 5b). For comparison, the observed volumetric uptake at 25 bar and 298 K is slightly lower than MOF-177 ($273 \text{ cm}^3 \text{ cm}^{-3}$),³² HKUST-1 ($276 \text{ cm}^3 \text{ cm}^{-3}$)³³ and Mg-MOF-74 (285 cm^3

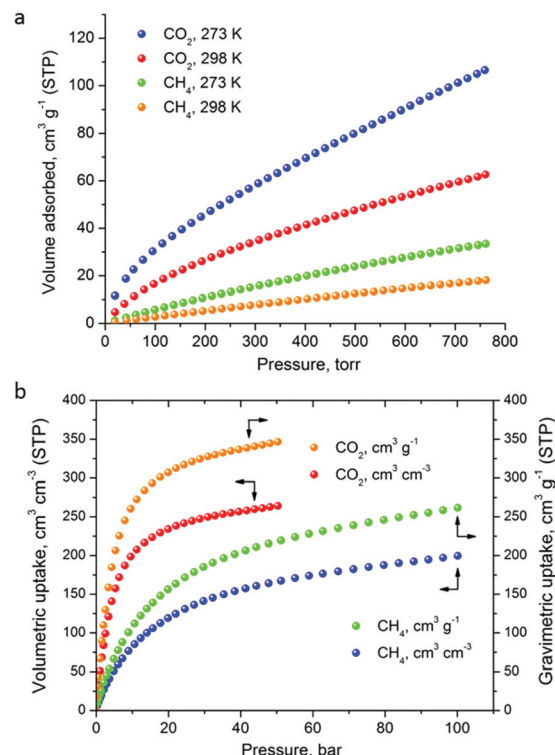


Fig. 5 (a) Low pressure CO_2 and CH_4 sorption isotherms of Cu-PEIP at the indicated temperatures, up to 1 bar. (b) The corresponding high pressure isotherms recorded at 298 K.

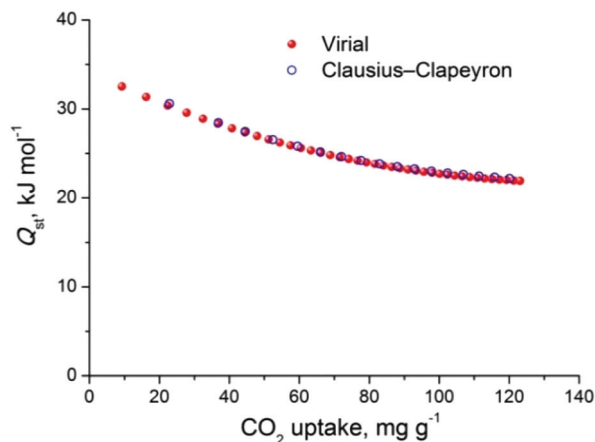


Fig. 6 CO_2 isosteric heat of adsorption (Q_{st}) in Cu-PEIP as a function of surface coverage, calculated from a virial-type analysis. The corresponding Clausius–Clapeyron calculation is also shown, for comparison.

cm^{-3})³⁴ but higher than MOF-5 ($225 \text{ cm}^3 \text{ cm}^{-3}$),³² gea-MOF-1 ($224 \text{ cm}^3 \text{ cm}^{-3}$),³⁵ MOF-210 ($127 \text{ cm}^3 \text{ cm}^{-3}$)³² and MOF-200 ($112 \text{ cm}^3 \text{ cm}^{-3}$).³²

The CH_4 adsorption isotherms up to 1 bar, shown in Fig. 5a revealed an uptake of $33.4 \text{ cm}^3 \text{ g}^{-1}$ (1.49 mmol g^{-1}) and $18.1 \text{ cm}^3 \text{ g}^{-1}$ (0.81 mmol g^{-1}) at 273 K and 298 K, respectively. From these isotherms, the calculated isosteric heat of adsorption, Q_{st} , at zero coverage using a virial-type equation, was



estimated to be 21.2 kJ mol^{-1} and remained nearly constant as a function of the surface coverage (Fig. S13 and S14†). Interestingly, this value is higher compared to HKUST-1 (17 kJ mol^{-1}) and among the highest reported for MOFs (Table S1†).^{36,37} For example, Ni-MOF-74 with a very high density of open metal sites showed a Q_{st} value of 21.4 kJ mol^{-1} .³⁶ Very recently, a novel MOF without open metal sites, denoted as MAF-38, showed a Q_{st} value of 21.6 kJ mol^{-1} at zero coverage.³⁸ Despite the high Q_{st} value for **Cu-PEIP**, a relatively high CO_2/CH_4 selectivity at low pressures, calculated using the IAST model, is observed reaching 8.5 at 273 K and 8.9 at 298 K.

High-pressure CH_4 measurements revealed that the total gravimetric ($\text{cm}^3 \text{ g}^{-1}$) and volumetric ($\text{cm}^3 \text{ cm}^{-3}$) uptake at 298 K, is 197 and 150 at 35 bar, 233 and 176 at 65 bar and 246 and 187 at 80 bar, respectively. The resulting gravimetric CH_4 working storage capacities in the pressure ranges 5–35 bar (US DOE standard),³⁹ 5–65 bar and 5–80 bar are estimated to be $131 \text{ cm}^3 \text{ g}^{-1}$, $167 \text{ cm}^3 \text{ g}^{-1}$ and $180 \text{ cm}^3 \text{ g}^{-1}$, respectively. The corresponding volumetric working capacities are $99 \text{ cm}^3 \text{ cm}^{-3}$, $125 \text{ cm}^3 \text{ cm}^{-3}$ and $136 \text{ cm}^3 \text{ cm}^{-3}$. These values are lower compared to the best performing MOFs (Tables S1 and S2 in the ESI†) due to the combination of high Q_{st} and a moderate surface area in **Cu-PEIP**. Compared to Ni-MOF-74, a MOF with a high Q_{st} for CH_4 and a slightly lower BET area ($1350 \text{ m}^2 \text{ g}^{-1}$), **Cu-PEIP** performs better in terms of the gravimetric working capacity at 5–65 bar (0.110 vs. 0.077 g g^{-1}) and 5–80 bar (0.129 vs. 0.091 g g^{-1}) (Tables S1 and S2 in the ESI†).

Low pressure, H_2 sorption isotherms recorded up to 1 bar at 77 K and 87 K revealed an uptake of $206 \text{ cm}^3 \text{ g}^{-1}$ ($2.06 \text{ wt}\%$) and $143 \text{ cm}^3 \text{ g}^{-1}$ ($1.43 \text{ wt}\%$), respectively, which is higher than the benchmark MOF-177 (142 and $82 \text{ cm}^3 \text{ g}^{-1}$) (Fig. S15†).⁴⁰ The calculated Q_{st} at zero coverage is 8.74 kJ mol^{-1} (Fig. S16 and S17†), which is significantly higher than some well-known and high performance MOFs⁴¹ including MOF-5 and MOF-177 (4.4 kJ mol^{-1}).⁴⁰ High pressure H_2 measurements, revealed an uptake of $614 \text{ cm}^3 \text{ g}^{-1}$ ($5.5 \text{ wt}\%$) which is lower than the ultra-high surface area MOFs such as MOF-210 and NU-100 (Fig. S18†). However, at 20 bar, due to a high Q_{st} , the uptake is $4.1 \text{ wt}\%$, comparable to the best performing MOFs.

Conclusions

Summarizing, a new Cu^{2+} MOF is reported that was isolated from the initial use of ligand PEIPH_2 in 3d MOF chemistry. Interestingly, **Cu-PEIP** displays novel structural features although pyridyl isophthalic acid tritopic ligands have been extensively employed in Cu MOF chemistry due to their ability to afford MOFs that display structural similarity with HKUST-1. The unique structural features of **Cu-PEIP** arose from the presence in its structure of both PEIPH_2 and $5\text{-NH}_2\text{-mBDCH}_2$ due to the capability of the tritopic Schiff base ligand to partially decompose in solution and appear in the reaction mixture together with its constituent moieties. After the structure of **Cu-PEIP** was known the synthesis of an analogous compound was targeted and achieved by employing a

rational synthetic procedure. The crystal structure of **Cu-PEIP** exhibits a 3D-framework that contains large pores ($\sim 9 \text{ \AA}$) and solvent accessible volume ($\sim 66.5\%$). In addition, it displays a unique trinodal underlying network consisting of pillared **kgm-a** layers and thus resembles the structure of HKUST-1. Gas sorption studies revealed that **Cu-PEIP** displays a significant BET area of $1785 \text{ m}^2 \text{ g}^{-1}$ and high CO_2 uptake, reaching 4.75 mmol g^{-1} at 273 K and 1 bar (2.80 mmol g^{-1} at 298 K) with good CO_2/CH_4 selectivity (8.5/8.9 at 273 K/298 K). Furthermore, high pressure gas sorption studies revealed that the volumetric CO_2 uptake at 25 bar ($242 \text{ cm}^3 \text{ cm}^{-3}$ at 298 K) is higher compared to representative high surface area MOFs, including MOF-5, MOF-200 and MOF-210. This study emphasizes the usefulness of the elements of serendipity introduced from the employment of polytopic Schiff base ligands in MOF chemistry for the synthesis of microporous MOFs with unique structural features and interesting sorption properties. Further investigations are in progress focusing on the isolation of **Cu-PEIP** analogues containing various functionalized derivatives of isophthalic acid and/or more elongated tritopic pyridyl isophthalic acid ligands.

Acknowledgements

This work was supported by the Cyprus Research Promotion Foundation Grant $\Delta\text{I}\Delta\text{AKT}\Omega\text{P}/0609/43$ which is co-funded by the Republic of Cyprus and the European Regional Development Fund. EM and AJT thank the University of Cyprus for an internal postdoctoral fellowship to EM.

Notes and references

- (a) M. Eddaoudi, D. B. Moler, H. Li, B. Chen, T. M. Reineke, M. O'Keeffe and O. M. Yaghi, *Acc. Chem. Res.*, 2001, **34**, 319–330; (b) D. Bradshaw, J. B. Claridge, E. J. Cussen, T. J. Prior and M. J. Rosseinsky, *Acc. Chem. Res.*, 2005, **38**, 273–282; (c) G. Ferey, *Chem. Soc. Rev.*, 2008, **37**, 191–214.
- (a) N. L. Rosi, J. Eckert, M. Eddaoudi, D. T. Vodak, J. Kim, M. O'Keeffe and O. M. Yaghi, *Science*, 2003, **300**, 1127–1129; (b) S. Zheng, T. Wu, J. Zhang, M. Chow, R. A. Nieto, P. Feng and X. H. Bu, *Angew. Chem., Int. Ed.*, 2010, **49**, 5362–5366; (c) B. F. Abrahams, M. J. Grannas, T. A. Hudson and R. Robson, *Angew. Chem., Int. Ed.*, 2010, **49**, 1087–1089.
- (a) H. Hayashi, A. P. Cote, H. Furukawa, M. O'Keeffe and O. M. Yaghi, *Nat. Mater.*, 2007, **6**, 501–506; (b) R. E. Morris and P. S. Wheatley, *Angew. Chem., Int. Ed.*, 2008, **47**, 4966–4981; (c) S. Noro, S. Kitagawa, M. Kondo and K. Seki, *Angew. Chem., Int. Ed.*, 2000, **39**, 2081–2084.
- (a) E. E. Moushi, T. C. Stamatatos, W. Wernsdorfer, V. Nastopoulos, G. Christou and A. J. Tasiopoulos, *Angew. Chem., Int. Ed.*, 2006, **45**, 7722–7725; (b) P. Dechambenoit and J. R. Long, *Chem. Soc. Rev.*, 2011, **40**, 3249–3265.



- 5 (a) L. Ma, J. M. Falkowski, C. Abney and W. Lin, *Nat. Chem.*, 2010, **2**, 838–846; (b) J.-S. Qin, D.-Y. Du, W. Guan, X.-J. Bo, Y.-F. Li, L.-P. Guo, Z.-M. Su, Y.-Y. Wang, Y.-Q. Lan and H.-C. Zhou, *J. Am. Chem. Soc.*, 2015, **137**, 7169–7177.
- 6 (a) Z. Hu, B. J. Deibert and J. Li, *Chem. Soc. Rev.*, 2014, **43**, 5815–5840; (b) A. Douvali, A. C. Tsipis, S. V. Eliseeva, S. Petoud, G. S. Papaefstathiou, C. D. Malliakas, I. Papadas, G. S. Armatas, I. Margiolaki, M. G. Kanatzidis, T. Lazarides and M. J. Manos, *Angew. Chem., Int. Ed.*, 2015, **54**, 1651–1656.
- 7 (a) O. M. Yaghi, H. Li, C. Davis, D. Richardson and T. L. Groy, *Acc. Chem. Res.*, 1998, **31**, 474–484; (b) M. Eddaoudi, J. Kim, D. Vodak, A. Sudik, J. Wachter, M. O’Keeffe and O. M. Yaghi, *Proc. Natl. Acad. Sci. U. S. A.*, 2002, **99**, 4900–4904.
- 8 (a) X.-L. Zhao and W.-Y. Sun, *CrystEngComm*, 2014, **16**, 3247–3258; (b) B. Manna, A. V. Desai and S. K. Ghosh, *Dalton Trans.*, 2016, **45**, 4060–4072.
- 9 (a) S. M. Cohen, *Chem. Rev.*, 2012, **112**, 970–1000; (b) O. Karagiari, W. Bury, J. E. Mondloch, J. T. Hupp and O. K. Farha, *Angew. Chem., Int. Ed.*, 2014, **53**, 4530–4540; (c) E. J. Kyprianidou, T. Lazarides, S. Kaziannis, C. Kosmidis, G. Itskos, M. J. Manos and A. J. Tasiopoulos, *J. Mater. Chem. A*, 2014, **2**, 5258–5266.
- 10 (a) N. Stock and S. Biswas, *Chem. Rev.*, 2012, **112**, 933–969; (b) Z.-J. Lin, J. Lü, M. Hong and R. Cao, *Chem. Soc. Rev.*, 2014, **43**, 5867–5895.
- 11 (a) C. N. R. Rao, S. Natarajan and R. Vaidhyanathan, *Angew. Chem., Int. Ed.*, 2004, **43**, 1466–1496; (b) E. E. Moushi, A. Kourtellaris, I. Spanopoulos, M. J. Manos, G. S. Papaefstathiou, P. N. Trikalitis and A. J. Tasiopoulos, *Cryst. Growth Des.*, 2015, **15**, 185–193.
- 12 (a) M.-S. Chen, Z.-S. Bai, T. Okamura, Z. Su, S.-S. Chen, W.-Y. Sun and N. Ueyama, *CrystEngComm*, 2010, **12**, 1935–1944; (b) X.-J. Deng, W. Gu, L. Wang, L.-F. Zeng and X. Z. Liu, *Z. Anorg. Allg. Chem.*, 2011, **637**, 708–712; (c) M.-S. Chen, M. Chen, S. Takamizawa, T. Okamura, J. Fan and W.-Y. Sun, *Chem. Commun.*, 2011, **47**, 3787–3789; (d) Y. Xiong, Y.-Z. Fan, R. Yang, S. Chen, M. Pan, J.-J. Jiang and C.-Y. Su, *Chem. Commun.*, 2014, **50**, 14631–14634; (e) Z. Chen, K. Adil, L. J. Weselinski, Y. Belmabkhout and M. Eddaoudi, *J. Mater. Chem. A*, 2015, **3**, 6276–6281.
- 13 (a) X. Zhang, Y.-Y. Huang and Y.-G. Yao, *Inorg. Chem. Commun.*, 2013, **28**, 49–51; (b) L. Qin, J.-S. Hu, L.-F. Huang, Y.-Z. Li, Z.-J. Guo and H.-G. Zheng, *Cryst. Growth Des.*, 2010, **10**, 4176–4183; (c) X. Zhang, J. Cheng, F. Chen, M. Sun and Y. Yao, *Inorg. Chem. Commun.*, 2011, **14**, 358–361; (d) M. C. Das and P. K. Bharadwaj, *J. Am. Chem. Soc.*, 2009, **131**, 10942–10949; (e) M. C. Das and P. K. Bharadwaj, *Chem. – Eur. J.*, 2010, **16**, 5070–5077; (f) A. Karmakar, L. M. D. R. S. Martins, S. Hazra, M. F. C. G. Da Silva and A. J. L. Pombeiro, *Cryst. Growth Des.*, 2016, **16**, 1837–1849; (g) M.-H. Xie, X.-L. Yang and C.-D. Wu, *Chem. – Eur. J.*, 2011, **17**, 11424–11427.
- 14 (a) V. Chandrasekhar, C. Mohapatra and R. Metre, *Cryst. Growth Des.*, 2013, **13**, 4607–4614; (b) V. Chandrasekhar and C. Mohapatra, *Cryst. Growth Des.*, 2013, **13**, 4655–4658.
- 15 S. S.-Y. Chui, S. M.-F. Lo, J. P. H. Charmant, A. G. Orpen and I. D. Williams, *Science*, 1999, **283**, 1148–1150.
- 16 (a) V. Guillermin, D. Kim, J. F. Eubank, R. Luebke, X. Liu, K. Adil, M. S. Lah and M. Eddaoudi, *Chem. Soc. Rev.*, 2014, **43**, 6141–6172; (b) J. F. Eubank, H. Mouttaki, A. J. Cairns, Y. Belmabkhout, L. Wojtas, R. Luebke, M. Alkordi and M. Eddaoudi, *J. Am. Chem. Soc.*, 2011, **133**, 14204–14207; (c) X. Liu, M. Oh and M. S. Lah, *Inorg. Chem.*, 2011, **50**, 5044–5053; (d) X. Liu, M. Oh and M. S. Lah, *Cryst. Growth Des.*, 2011, **11**, 5064–5071; (e) S. Xiang, J. Huang, L. Li, J. Zhang, L. Jiang, X. Kuang and C.-Y. Su, *Inorg. Chem.*, 2011, **50**, 1743–1748; (f) B. Moulton, J. Lu, R. Hajndl, S. Hariharan and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2002, **41**, 2821–2824; (g) S. A. Bourne, J. Lu, A. Mondal, B. Moulton and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2001, **40**, 2111–2113.
- 17 (a) E. J. Kyprianidou, G. S. Papaefstathiou, M. J. Manos and A. J. Tasiopoulos, *CrystEngComm*, 2012, **14**, 8368–8373; (b) M. J. Manos, E. J. Kyprianidou, G. S. Papaefstathiou and A. J. Tasiopoulos, *Inorg. Chem.*, 2012, **51**, 6308–6314; (c) C. G. Efthymiou, E. J. Kyprianidou, C. J. Milios, M. J. Manos and A. J. Tasiopoulos, *J. Mater. Chem. A*, 2013, **1**, 5061–5069.
- 18 Oxford Diffraction, *CrysAlis CCD and CrysAlis RED*, Oxford Diffraction Ltd., Abingdon, UK, 2008.
- 19 M. C. Burla, R. Caliendo, M. Camalli, B. Carrozzini, G. L. Casciarano, L. De Caro, C. Giacovazzo, G. Polidori and R. Spagna, *J. Appl. Crystallogr.*, 2005, **38**, 381–388.
- 20 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Fundam. Crystallogr.*, 2008, **64**, 112–122.
- 21 L. J. Farrugia, *J. Appl. Crystallogr.*, 1999, **32**, 837–838.
- 22 K. Brandenburg, *DIAMOND, Version 2003.2001d*, Crystal Impact GbR: Bonn, Germany, 2006.
- 23 <http://www.ccp14.ac.uk/ccp/web-mirrors/x-seed/>.
- 24 A. L. Spek, *J. Appl. Crystallogr.*, 2003, **36**, 7–13.
- 25 C. F. Macrae, P. R. Edgington, P. McCabe, E. Pidcock, G. P. Shields, R. Taylor, M. Towler and J. Van De Streek, *J. Appl. Crystallogr.*, 2006, **39**, 453–457.
- 26 L. Sarkisov and A. Harrison, *Mol. Simul.*, 2011, **37**, 1248–1257.
- 27 Y. Lin, C. Kong and L. Chen, *RSC Adv.*, 2016, **6**, 32598–32614.
- 28 Y. Lin, C. Kong and L. Chen, *RSC Adv.*, 2012, **2**, 6417–6419.
- 29 R. Vaidhyanathan, S. S. Iremonger, K. W. Dawson and G. K. H. Shimizu, *Chem. Commun.*, 2009, 5230–5232.
- 30 Y. Fu, D. Sun, Y. Chen, R. Huang, Z. Ding, X. Fu and Z. Li, *Angew. Chem., Int. Ed.*, 2012, **51**, 3364–3367.
- 31 J. Ethiraj, E. Albanese, B. Civalleri, J. G. Vitillo, F. Bonino, S. Chavan, G. C. Shearer, K. P. Lillerud and S. Bordiga, *ChemSusChem*, 2014, **7**, 3382–3388.
- 32 H. Furukawa, N. Ko, Y. B. Go, N. Aratani, S. B. Choi, E. Choi, A. Ö. Yazaydin, R. Q. Snurr, M. O’Keeffe, J. Kim and O. M. Yaghi, *Science*, 2010, **329**, 424–428.



- 33 J. Moellmer, A. Moeller, F. Dreisbach, R. Glaeser and R. Staudt, *Microporous Mesoporous Mater.*, 2011, **138**, 140–148.
- 34 J. M. Simmons, H. Wu, W. Zhou and T. Yildirim, *Energy Environ. Sci.*, 2011, **4**, 2177–2185.
- 35 V. Guillermin, Ł. J. Weseliński, Y. Belmabkhout, A. J. Cairns, V. D'Elia, Ł. Wojtas, K. Adil and M. Eddaoudi, *Nat. Chem.*, 2014, **6**, 673–680.
- 36 Y. He, W. Zhou, G. Qian and B. Chen, *Chem. Soc. Rev.*, 2014, **43**, 5657–5678.
- 37 I. Spanopoulos, C. Tsangarakis, E. Klontzas, E. Tylianakis, G. Froudakis, K. Adil, Y. Belmabkhout, M. Eddaoudi and P. N. Trikalitis, *J. Am. Chem. Soc.*, 2016, **138**, 1568–1574.
- 38 J.-M. Lin, C.-T. He, Y. Liu, P.-Q. Liao, D.-D. Zhou, J.-P. Zhang and X.-M. Chen, *Angew. Chem., Int. Ed.*, 2016, **55**, 4674–4678.
- 39 Y. Peng, V. Krungleviciute, I. Eryazici, J. T. Hupp, O. K. Farha and T. Yildirim, *J. Am. Chem. Soc.*, 2013, **135**, 11887–11894.
- 40 H. Furukawa, M. A. Miller and O. M. Yaghi, *J. Mater. Chem.*, 2007, **17**, 3197–3204.
- 41 D. P. Broom, C. J. Webb, K. E. Hurst, P. A. Parilla, T. Gennett, C. M. Brown, R. Zacharia, E. Tylianakis, E. Klontzas, G. E. Froudakis, T. A. Steriotis, P. N. Trikalitis, D. L. Anton, B. Hardy, D. Tamburello, C. Corgnale, B. A. Van Hassel, D. Cossement, R. Chahine and M. Hirscher, *Appl. Phys. A*, 2016, **122**, 151.

