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## Dynamic transformation of active sites in energy and environmental catalysis

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Active sites play a pivotal role in photo/electrocatalysis, particularly in the transition from fossil fuels to clean, efficient and renewable energy sources. Precise identification of catalyst active sites and understanding of their dynamic transformation are crucial for engineering the activity, selectivity and stability of a catalyst for a specific reaction. Herein, we provide an in-depth and interdisciplinary overview of the recent advancements in dynamic transformation of active sites in photo/electrocatalysis. Firstly, we explore the underlying principles of the dynamic reconstruction, focusing on dynamic transformations in surface structure, composition and properties. Subsequently, advanced *operando/in situ* characterization for dynamic transformation is summarized, to provide mechanistic insights for the identification of such processes. In order to improve catalytic performance, we discussed comparatively the triggers and the corresponding reaction mechanisms of the dynamic process. Finally, we present an insightful analysis of the challenges and the future prospects for the applications of dynamic transformation of active sites in photo/electrocatalysis.

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### Broader context

In recent years, there has been a surge in research focused on understanding the dynamic transformations of active sites in photo/electrocatalysis, driven by the urgent need for transition from fossil fuels to renewable energy sources. Our review provides an interdisciplinary overview of recent advancements in the area of energy and environmental catalysis, encompassing the origins, identification methods, and targeted construction of dynamic transformations. We commence our review by exploring the origins of dynamic reconstruction under reaction conditions, delving into dynamic transformations involving changes in surface properties, composition, and structure. Subsequently, we summarize advanced *operando/in situ* characterization for dynamic transformations, offering valuable insights for the identification and detailed investigation of these processes. To manipulate the dynamic conversion of active sites and improve catalytic performance, we discuss various triggers, corresponding reaction mechanisms, and the resulting photo/electrocatalytic performance. Additionally, we present an insightful analysis of the challenges and outline future prospects for the dynamic transformation of active sites in photo/electrocatalysis.

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# 1. Introduction

In the contemporary epoch, the burgeoning severity of global energy and environmental crises has considerably expedited the development of renewable energy technologies, encompassing notably hydroelectric, wind, and solar power.<sup>1–3</sup> Among these, solar energy has led the way to the development of sustainable energy storage and conversion systems, turning its instability and intermittency to continuous and reliable utilization, *e.g.* via photovoltaics and photoelectrochemical catalysis.<sup>4,5</sup> Meanwhile, electrochemistry has drawn great scientific interest for providing sustainable energy carriers, such as hydrogen (H<sub>2</sub>), as well as energy conversion and storage devices, including metal–air/ion batteries, electrolyzers and fuel cells.<sup>6,7</sup> Such systems rely on various photo/electrocatalytic redox couples, such as the oxygen reduction reaction (ORR), oxygen evolution reaction (OER), hydrogen evolution reaction (HER), carbon dioxide reduction reaction (CO<sub>2</sub>RR), nitrogen reduction reaction (NRR) and NO<sub>x</sub> reduction reaction (NO<sub>x</sub>RR).<sup>8–11</sup> Catalysts play a critical role in activating inert reactants and determining reaction pathways, which are often influenced by the dynamic transformation of the active sites that adapt to the chemical potential environment under operational conditions.<sup>12,13</sup> Therefore, comprehending the dynamic behavior of active sites is essential for the rational design of catalysts to ensure high-efficiency energy and environmental applications.

Since Taylor introduced the concept of the ‘active site’ in 1925,<sup>14</sup> research on fundamental catalysis has focused on the identification of catalytically active species and the comprehensive understanding of their operational mechanisms.<sup>15</sup> Catalytically active sites, representing a small proportion of a catalyst’s energetically favorable surface, comprise individual atoms or specific groups of atoms concentrated at surface facets and defects, such as edges, corners and amorphous regions.<sup>16</sup> Generally, catalysts lower the activation energy for molecular conversion by chemically binding reactant molecules, thereby leading to proton and/or electron transfer between the reactants and the active sites.<sup>17</sup> Therefore, catalytic

processes, particularly those that necessitate energy-driven conditions, such as temperature, voltage or photo-irradiation, inevitably induce changes in the geometric and electronic structures of catalysts. These changes, referred to as dynamic reconstruction, encompass alterations in chemical valences, phases, structures, and coordination environments.<sup>18,19</sup> For thermal catalysis, the charge transfer primarily occurs on the catalyst’s surface and can be attributed to surface reconstruction resulting from interactions with reactants.<sup>20</sup> However, electrocatalytic and photocatalytic processes generally involve charge transfer throughout the entire catalyst rather than being confined to surface active sites.<sup>21</sup> Specifically, the absorption of photons triggers the generation of charge carriers in photocatalysts, and the application of an external voltage drives electron migration towards electrodes and surfaces during electrocatalysis.<sup>22</sup> Active sites in photo(electro)catalysts are thermodynamically metastable and transiently arise from interactions with, under photo illumination and/or applied bias, electrolytes and reactants/intermediates.<sup>23</sup> By investigating the dynamic transformation of catalysts, valuable information about active sites can be obtained, allowing us to uncover possible mechanisms for the activation and deactivation of (pre)catalysts.

The increasing global focus on sustainability has prompted the substitution of conventional thermal heterogeneous catalytic processes with renewable-powered photo/electrocatalytic processes.<sup>24</sup> Fig. 1 showcases the developments and discoveries of the reconstruction of various chemical compounds and materials. We observe a trend towards increasingly precise construction of such materials, transitioning from the elucidation of oxide chemical reconstructions to the precise modulation of reconstructions involving specific elements or individual atoms. This progression underscores notable advancements in our comprehension and characterization of the reconstruction mechanisms. In recent years, the development of *in situ/operando* characterization techniques has unveiled substantial dynamic transformation in catalysts during photo/electrocatalytic reactions.<sup>25</sup> The dynamic transformations of catalysts span from atomic-scale surface property

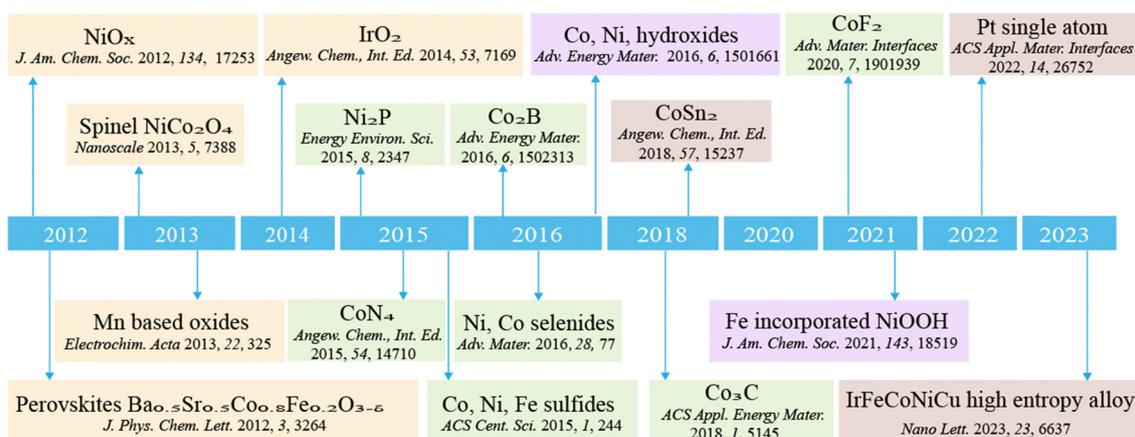


Fig. 1 Chronogram of the developments in reconstructed catalysts.



changes (a few Å) to phase transformations at a relatively larger scale (tens of nm), and even significant variations in morphologies (to mm). As illustrated in the summary plot of the relationship between the size of pre-catalysts and the thickness of reconstruction layers (Fig. 2), it is apparent that specific elements or nanostructures frequently correspond to varying scales of reconstruction. This phenomenon underscores our ability to strategically design catalysts with tailored reconstruction patterns. Generally speaking, these dynamic transformations are able to modulate photo/electrocatalytic behaviors, such as adsorption, activation and desorption, and can be engineered to enhance catalytic performance.<sup>26,27</sup> However, it should also be acknowledged that the reconstructed photo/electrocatalysts may experience ongoing restructuring throughout a catalytic process, which may result in the loss of active components and the degradation in catalytic activity.<sup>28</sup> This dynamic transformation also serves as a clear indication of the eventual deterioration of the catalyst, commonly referred to as catalyst corrosion.<sup>29,30</sup> In this regard, optimizing the reconstruction process to produce a profusion of active sites with high intrinsic activity represents a powerful strategy for enhancing the catalytic performance of photo/electrocatalysts.<sup>31</sup>

To date, the existing reviews on the dynamic transformation of electrocatalysts mainly focus on either one or a limited number of energy and environmental applications, such as the OER, CO<sub>2</sub>RR and HER.<sup>32</sup> In addition, they enumerate various types of restructuring catalysts without associating restructuring triggers with specific types of elements.<sup>33,34</sup> It is

highly desirable to rationalize the discovered correlations between restructuring triggers and different categories of reported elements for engineering the reconstruction from various perspectives (Fig. 3). This endeavor aims to facilitate more effective utilization of certain elements in constructing specific catalysts *via* restructuring processes. Recent studies on dynamic transformation have been found in diverse photo/electrocatalytic systems for energy and environmental applications, albeit with varied quantity of literature available across these fields.<sup>35</sup> The emergence and maturity of nanotechnology have dissolved the boundaries of single-disciplinary fields and facilitated multidisciplinary investigations across diverse scientific domains.<sup>36</sup> Undoubtedly, understanding the dynamic transformation mechanism of photo/electrocatalysts, optimizing the reconstruction processes, and uncovering the true structure–activity relationship call for interdisciplinary efforts in order to achieve a noteworthy breakthrough in formulating photo/electrocatalysts.<sup>37</sup> In addition, this can form a robust foundation for comprehending the dynamic transformation of catalysts and serve as a pioneering reference in engineering multifunctional catalysts for photochemical and electrochemical reactions.

This review aims to present an in-depth analysis, critical assessment and new insight of the dynamic transformation into active sites in the realm of energy and environmental catalysis, including the origin and types of dynamic transformation, the underlying principles, the identification methods with related *operando/in situ* or multiangle/level analyses, and

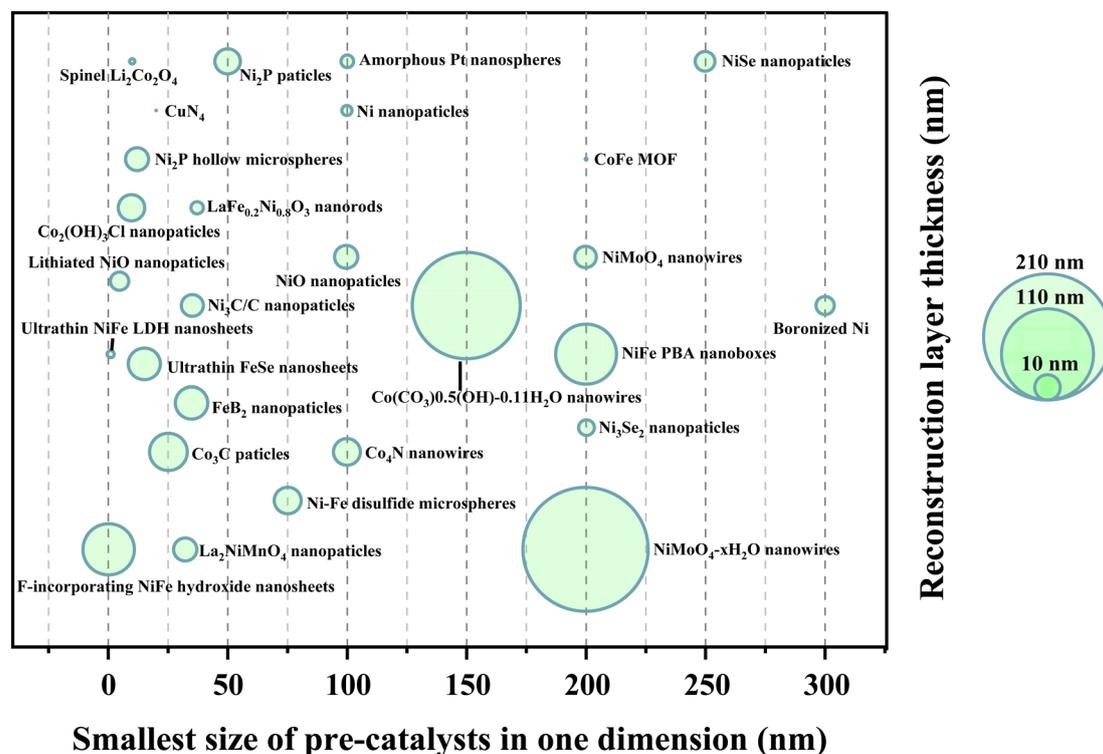


Fig. 2 Scatter plot of the relationship between the size of pre-catalysts and the thickness of reconstruction layers.



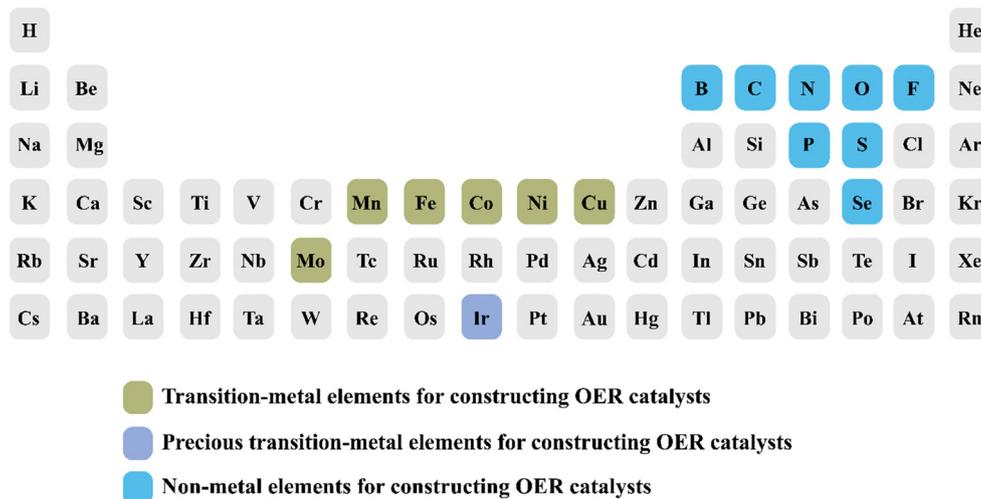


Fig. 3 Reported elements for constructing transition metal-based OER catalysts.

several triggers of dynamic transformation, which are categorized into single atom catalyst induction, defect engineering, heterostructure construction, leaching, and crystallinity modification, as well as the photo/electrocatalytic applications in energy and environmental domains, such as electrocatalytic HER/OER/CO<sub>2</sub>RR/NRR and photocatalytic pollutant degradation/HER/OER/CO<sub>2</sub>RR. Note that the focus is on the theory-structure-activity relationships and some optimization strategies of dynamic transformation, benefiting from the mitigation of the adverse effects resulting from these transformations and exploring potential positive effects that may emerge. Finally, we propose the existing challenges and novel opportunities that hold the promise of propelling the development of dynamic transformation in photo/electrocatalytic reactions and contribute to the rational design of high-performance earth-abundant photo/electrocatalysts.

## 2. Fundamentals of dynamic transformation

Catalytic systems involving dynamic transformation of active sites have emerged as a common occurrence in the scientific literature, garnering increasing attention and interest from scholars across various disciplines.<sup>38</sup> An in-depth study of dynamic transformation enables precise customization of intrinsic properties and facilitates the discovery of new phenomena in nanomaterials. Theoretical insights into dynamic transformation are crucial for understanding reaction mechanisms and optimizing catalyst design for specific processes. The purpose of this section is to deliver a thorough introduction to the fundamentals of the dynamic transformation of photo/electrocatalysts, encompassing their origin, various types and identification methods (Fig. 4).

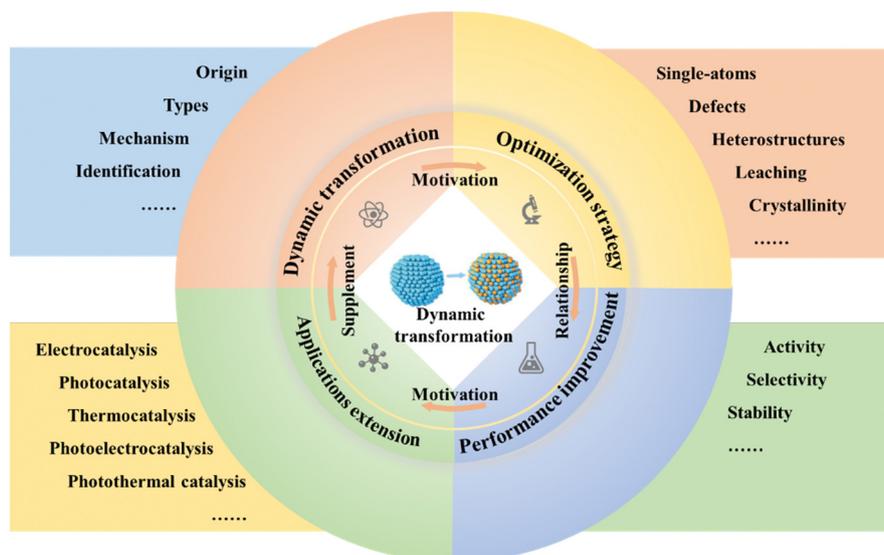


Fig. 4 Summary of the correlation of various factors that can affect the dynamic transformation during energy and environmental catalysis.



## 2.1 Origins

To gain a deeper understanding of the dynamic evolution of active sites, it is crucial to trace their origin. It is widely acknowledged that the realization of high catalytic activity and selectivity in most catalytic reactions critically hinges upon the meticulous establishment of tailored reaction conditions. However, such stringent conditions may inevitably induce the dynamic structural evolution of catalysts.<sup>39</sup> Generally, the energy input intensity (*e.g.*, applied bias and light radiation) serves as the driving force to induce the dynamic transformation of photo/electrocatalysis.<sup>11</sup> Variable testing conditions, including reaction medium, temperature, and reaction time, play a crucial role in modulating the energy input intensity, which can influence the stability of precatalysts and impact their degree of reconstruction.<sup>40</sup>

**2.1.1 Electrocatalytic reactions.** In an electrocatalytic reaction, the applied potential acts as the driving force necessary to surpass the minimum Gibbs free energy barrier for the reaction.<sup>41</sup> Under practical operations, unsatisfactory reaction conditions and substantial kinetic barriers often require the application of a considerably more negative or positive potential than the equilibrium potential to enable practical reactant conversions with improved scalability.<sup>42</sup> Extensive research has shown that the applied potentials used to drive molecule conversions often overlap to some extent with the redox potentials of the electrocatalysts.<sup>43</sup> When the applied potential exceeds the redox potential of the electrocatalysts, the conversion of reactant molecules and the variation of the surface properties of electrocatalysts occur simultaneously.<sup>44</sup> If the process remains irreversible despite returning the potential to its original state, it will drive persistent evolution, leading to the gradual emergence of a newly reconstructed layer with different species.<sup>45</sup>

In order to elucidate the electrocatalyst reconstruction behavior, electrocatalysts can be conceptually simplified as materials consisting of active sites and ligands. The active center, often composed of a metallic or nonmetallic element (*e.g.*, carbon), exhibits electrocatalytic activity, while the ligands can encompass either metallic/nonmetallic elements or molecules capable of coordination. Then, the dynamic transformation of the electrocatalyst can be perceived as encompassing the oxidation/reduction and dissolution/redeposition of active sites and ligands, along with the incorporation of new ligands. Typically, electrocatalysts tend to undergo oxidation or partial oxidation when the applied potential surpasses the theoretical potential for electron loss from active sites or ligands, resulting in the dissolution and substitution of active sites or ligands. In contrast, at sub-theoretical electrode potentials for electron acquisition, electrocatalysts experience reduction or partial reduction. This can lead to ligand dissolution and substitution, but active site dissolution is generally not observed in this case. During potential cycling, active sites/ligands that have dissolved at high potential are subsequently redeposited onto the surface of the electrocatalyst when the potential is reduced.

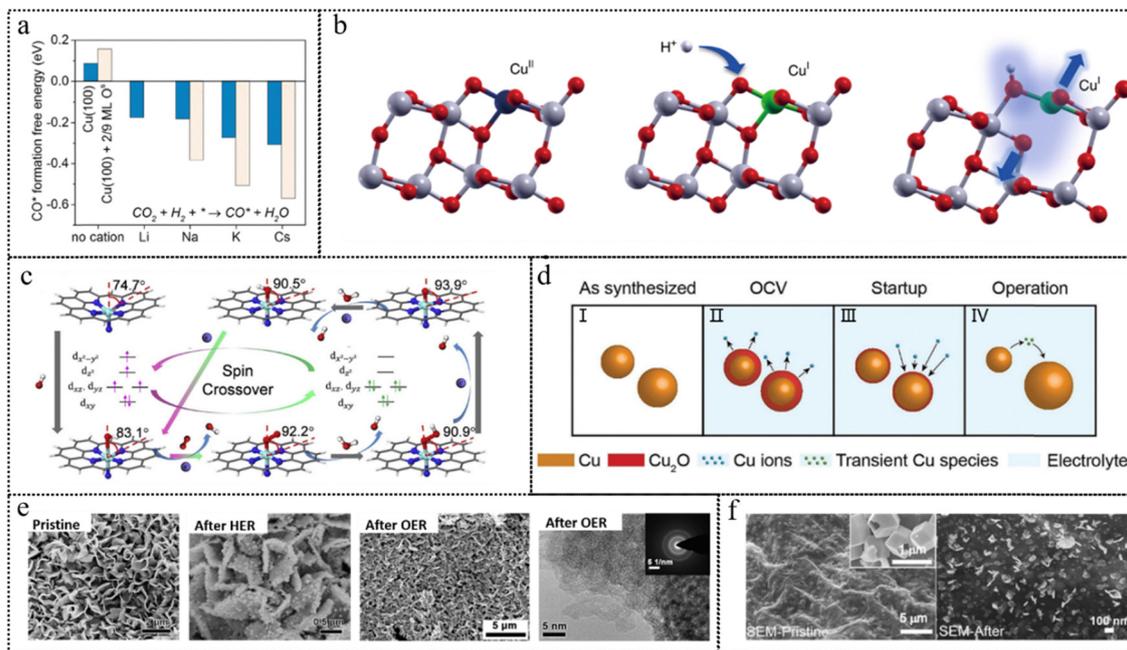
The testing conditions can primarily influence specific redox potentials and govern the extent of electrocatalyst

reconstruction. Under specific testing conditions (temperature, electrolyte and pressure), the reduction of initial potential affords the creation of highly reactive sites at low potential, thereby decreasing the overpotential of the electrocatalytic reaction. Among these testing conditions, the pH value and/or cation size of electrolytes have a large influence on the dynamic reconstruction of electrocatalysts. In the context of ligand substitution reconstruction, oxygen acts as a potential substituent within an acidic environment, while in an alkaline medium, a hydroxyl group could potentially undergo replacement. Furthermore, the cation sizes in the electrolyte exhibit a pronounced influence on the adsorption characteristics of the electrocatalyst. For example, through the synergistic effect of enlarging the alkali-metal cationic sizes within the electrolytic medium and incorporating subsurface oxygen species that promote the adsorption for CO<sub>2</sub>, a notable enhancement in the C–C coupling process on the reconstructed CuO<sub>x</sub> electrodes was observed (Fig. 5a).<sup>46</sup>

**2.1.2 Photocatalytic reactions.** During photocatalytic reactions, the photon energy triggers the generation and separation of photoinduced electrons and holes, and these charge carriers then migrate to the surface active sites to initiate the redox reactions. To compensate for the loss of charge carriers during migration, it is often necessary to increase the intensity of the incident light radiation beyond the minimum required value.<sup>52</sup> This ensures an adequate supply of charge carriers to sustain and drive the photocatalytic reactions effectively. Another fact that cannot be ignored is that the interaction between these charge carriers and the semiconductor and/or co-catalyst upon contact is inevitable, resulting in the occurrence of oxidation and reduction processes of the photocatalyst induced by photoinduced electrons and holes, respectively.<sup>53</sup> It is noteworthy that the incorporation of specific sacrificial agents can consume a certain quantity of photoinduced carriers, thereby diminishing the efficacy of photocatalyst reconstruction. This suggests that the degree of photocatalyst reconstruction can be modulated by the type and concentration of sacrificial agents.

Beyond the photoinduced carriers, some reactants can also participate in the reconstruction of the photocatalysts. The interaction between the reactants and the photocatalyst surface can lead to the formation of new active sites, enhance the charge separation and transfer efficiency, and ultimately improve the photocatalytic performance of the system. For example, the surface of TiO<sub>2</sub> can undergo atomic rearrangement when exposed to H<sub>2</sub>O molecules, and light irradiation further amplifies the extent of these structural modifications.<sup>54</sup> Thin amorphous layers were formed, measuring approximately 1–2 atomic layers in thickness, on crystalline TiO<sub>2</sub> nanoparticles subsequent to their exposure to ultraviolet irradiation and H<sub>2</sub>O vapor. In addition, molecular co-catalysts that exhibit strong affinity towards photocatalyst surfaces have shown remarkable potential in inducing electronic structural modifications, thus resulting in a mild surface reconstruction. The utilization of molecular co-catalysts, such as CF<sub>3</sub>CO<sub>2</sub>H and ethylenediamine, has exhibited commendable enhancement in photocatalytic performance.<sup>55,56</sup>





**Fig. 5** (a)  $\text{CO}^*$  formation free energy calculated in the absence and presence of  $\text{Li}^*$ ,  $\text{Na}^*$ ,  $\text{K}^*$ , and  $\text{Cs}^*$  on bare  $\text{Cu}(100)$  and  $\text{Cu}(100)$  with  $2/9 \text{ ML O}^*$ . This figure has been reproduced from ref. 46 with permission from American Chemical Society, copyright 2018. (b) Schematic representation of the variation in the chemical state of Cu during photocatalysis. This figure has been reproduced from ref. 47 with permission from Nature, copyright 2019. (c) Structure and dynamics of the  $\text{N}-\text{FeN}_4\text{C}_{10}$  moiety in the ORR with optimized geometries (the angles represent the deviation of iron from the  $\text{N}_4$ -plane). This figure has been reproduced from ref. 48 with permission from Elsevier, copyright 2020. (d) Schematic representation of the catalyst reconstruction process: (I) the as-synthesized metallic Cu NPs, (II) the catalyst oxidation and dissolution of Cu ions into the electrolyte, (III) the redeposition of dissolved ions onto the catalyst when a reducing potential is applied, and (IV) the dissolution of smaller particles into transient species, whose nature remains unknown to date, that contribute to the growth of larger nanoparticles under a constant applied potential during cell operation. This figure has been reproduced from ref. 49 with permission from Wiley-VCH, copyright 2020. (e) SEM images and a high-resolution TEM image of pristine FeS/IF, FeS/IF after the HER and FeS/IF after the OER. This figure has been reproduced from ref. 50 with permission from Elsevier, copyright 2018. (f) SEM images of the as-prepared (pristine)  $\text{CoS}_x/\text{NGF}$  and the  $\text{CoS}_x/\text{NGF}$  after the OER. This figure has been reproduced from ref. 51 with permission from American Chemical Society, copyright 2018.

## 2.2 Types

The dynamic transformations of photo/electrocatalysts can be systematically classified based on the length scale of these transformations, spanning a diverse range that extends from the atomic-level alterations (a few Å) manifesting in the surface properties, to the macroscopic dimensions (tens of nm) associated with phase transformations, and even encompassing obvious structural reconstructions. The comprehensive elucidation is presented as follows.

### 2.2.1 Surface properties

**2.2.1.1 Chemical states.** The surface properties of photo/electrocatalysts can transform as a consequence of intricate interactions among reactants, charge carriers and active sites. The chemical states of a catalyst can readily undergo alterations during the process of photo/electrocatalysis.<sup>57</sup> Typically, the occurrence of more negative potential conditions and the generation of photogenerated electrons tend to facilitate the establishment of a metallic state. While exposed to relatively positive potentials or in contact with photogenerated holes, catalysts generally exhibit a preference for existing in an oxidized state.<sup>58</sup> It is easily comprehensible that the initially present oxidized species of metals, such as titanium oxide or copper oxide, in catalysts can undergo reduction to a lower

valence state by photogenerated electrons or by the application of cathodic potentials.<sup>59</sup> It is crucial to emphasize that even in the case of a metal-based initial catalyst, rapid oxidation reactions can occur upon exposure to air or the electrolyte under open circuit potential conditions, resulting in the formation of a thin oxide layer on the surface of the metal catalyst.<sup>60</sup> This oxide layer can also be reduced by photogenerated electrons or under cathodic potentials. The presence and characteristics of this reduced oxide layer have a pronounced influence on subsequent catalytic processes.<sup>61</sup> One notable example is the reduction of  $\text{Cu}_2\text{O}$  to metallic Cu observed under  $\text{CO}_2\text{RR}$  conditions.<sup>62</sup> This reduction process results in an increased metallicity of the copper electrode, which subsequently promotes C–C bond formation under high cathodic polarization. In addition, the PtO species of the PtO– $\text{TiO}_2$  photocatalyst experienced partial reduction when exposed to light irradiation due to the presence of photogenerated electrons.<sup>63</sup>

The dynamic transformation of catalysts to an oxidized state generally occurs in oxidation reactions, which necessitate a significant driving force to overcome the slow kinetic barriers, such as the OER.<sup>64</sup> Metal redox in oxide-hydroxide catalysts is feasible due to the mobility of lattice oxygenic ligands, enabling



their exchange with water molecules.<sup>65</sup> For example, a significant proportion of Co ions undergoes a valence state transition from Co<sup>3+</sup> to Co<sup>4+</sup> in response to voltage and time variations accompanied by spontaneous delithiation.<sup>66</sup> The high OER activity is primarily attributed to the highly oxidized Co<sup>4+</sup> site, as opposed to the Co<sup>3+</sup> site or the oxygen vacancy site. NiO<sub>x</sub> species, commonly used as oxidation cocatalysts for photocatalytic water splitting, can be oxidized to form NiO(OH) when exposed to photogenerated holes during the early stages of the reaction.<sup>67</sup> The formation of Ni<sup>3+</sup> species played a crucial role in facilitating the OER process and contributed to the overall efficiency of the photocatalytic water-splitting reaction. Beyond metal oxides, high-valent metals have been found to be crucial in various OER catalyst systems, including metal phosphides, sulfides, phosphates, carbides, *etc.* Starting from cobalt phosphates, Co<sup>2+</sup> in the pristine precatalyst can be oxidized to Co<sup>3+</sup>, which acts as the active species for the OER activity.<sup>68,69</sup>

**2.2.1.2 Coordination environments.** In empirical investigations and utilizing powdered samples, it has been substantiated that altering the spatial positioning of an individual metal atom on a support surface can result in a discernible change in the coordination environment surrounding the metal atom. The mobility of catalytic metal atoms across a surface or the removal of coordinated anions in the support, thereby creating a vacancy, can potentially lead to the emergence of an alternative coordination environment. The ensuing alterations in both geometric and electronic structures exert a significant influence on the catalytic activity. Taking the Cu-TiO<sub>2</sub> catalyst composed of isolated Cu<sup>II</sup> sites as a typical example, the presence of photogenerated electrons leads to the formation of Cu<sup>I</sup> species encircled by a distorted support structure, arising from the partial reduction of Ti<sup>IV</sup> to Ti<sup>III</sup> (Fig. 5b). Such modified Cu<sup>I</sup> species serve as the active sites responsible for the photocatalytic HER.<sup>47</sup>

During the ORR process, there is often a dynamic transformation observed between planar and curved coordination geometries in the structural evolution of M-N<sub>x</sub> sites. Near the ORR onset potential (0.9 V *vs.* RHE), a reduction in the Fe-N bond length occurs as the central Fe atom in N-FeN<sub>4</sub>C<sub>10</sub> moves towards the N<sub>4</sub>-plane (Fig. 5c).<sup>48</sup> This structural change was coupled with the transformation of Fe<sup>2+</sup> from a high-spin (HS) to a low-spin (LS) electronic state with the formation of an O<sub>2</sub>-FeN<sub>5</sub> intermediate. Conversely, during the formation of the O<sub>2</sub>-FeN<sub>4</sub> intermediate at higher bias potentials (0.7 V and 0.5 V *versus* RHE), as the central Fe atom within the FeN<sub>4</sub>C<sub>12</sub> species shifted away from the N<sub>4</sub>-plane, the Fe-N bond length underwent an expansion. This structural modification was concurrent with the conversion of the Fe<sup>2+</sup> electronic state from LS to HS.

## 2.2.2 Composition changes

**2.2.2.1 Element redistribution.** Element redistribution is a prevalent occurrence in the reactions of bimetallic or multi-metallic catalysts, and one illustrative example of this phenomenon is found in alloy catalysts. Alloy nanoparticles have garnered widespread utilization in a multitude of

electrocatalytic reactions, encompassing the ORR, alcohol oxidation and water oxidation. Notably, the distribution of metals within a catalyst after utilization can exhibit marked disparities when compared to the initial pristine or precatalyst state. For instance, PtM nanoparticles, where M typically represents first-row transition metals like Fe, Co, Ni, or Cu, undergo a transformation into core-shell structures during electrochemical processes, characterized by a PtM core enveloped by a Pt shell, owing to the electrochemical de-alloying. The de-alloyed bimetallic nanoparticles exhibit enhanced activity in comparison to their more homogeneous counterparts, which can be attributed to the distinctive electronic properties of the Pt shell, as well as the lattice strain that arises at the interface between the Pt shell and the alloyed core.<sup>70</sup>

The phenomenon of segregation in composite metal-oxide materials leads to the transformation of a uniform solid into one characterized by the presence of interfacial structures.<sup>71</sup> This can be exemplified by considering perovskite materials, where certain metal ions have the propensity to undergo reduction during the process of high-temperature solid oxide fuel cells, resulting in the formation of metallic regions.<sup>72</sup> In most reported works, the active catalyst is not a pristine starting (pre)catalyst but, instead, an *in situ*-formed structure. In some cases, chemical segregation can yield an active material, but it may also exert a detrimental influence on electrocatalytic performance. Taking MnCo spinel oxides as an example,<sup>73</sup> Mn sites exhibited the ability to activate O<sub>2</sub>, while Co sites are proficient in activating H<sub>2</sub>O, in the alkaline ORR. However, when Mn or Co oxides were segregated into distinct surface patches, the cooperative interaction between these two different metals was compromised, resulting in an inefficient performance of the ORR.

**2.2.2.2 Phase transformation.** Under most circumstances, a pristine (pre)catalyst undergoes a crystal phase transformation when exposed to the catalytic reaction conditions. This transformation can involve transitioning into an amorphous phase or evolving into different chemical substances. Subsequently, the catalyst adopts a new crystal phase, enabling its continued participation in the catalytic reaction. Metal-organic frameworks (MOFs) represent a category of compounds characterized by the formation of coordination bonds between organic ligands and metal ions or clusters. Within MOFs, these metal components serve as active sites, exhibiting notable attributes such as high mono-dispersibility and atomic efficiency. Nonetheless, an inherent issue with MOF-based photocatalysts is their vulnerability to instability under reaction conditions. The presence of photogenerated electrons, holes, and radicals can lead to the degradation or collapse of the framework structures of MOFs. Accordingly, the application of MOF photocatalysts has frequently been associated with a decrease in crystallinity or a transition to amorphous structures.<sup>74</sup> Several MOFs, such as NiCo-MOF, NiFe-MOF, and CoFe-MOF, have risen in prominence for their pivotal roles in electrochemical reactions, notably as effective electrocatalysts for the OER. Recent investigations have confirmed that the actual catalytic species for the



OER in MOFs are derived from the reconstruction of oxyhydroxides rather than originating from the pristine metal nodes. A ligand substitution mechanism has been proposed to elucidate the reconstruction of ZIF-67 during the OER.<sup>75</sup> Upon exposure to an electrolyte containing alkaline hydroxide ions, the tetrahedral Co-bonded mIM molecules on the surface layer of ZIF-67 would be substituted by hydroxide ions, resulting in the formation of  $\text{Co(mIM)}_3\text{OH}$ ,  $\text{Co(mIM)}_2\text{OH}_2$ ,  $\text{Co(mIM)}\text{OH}_3^-$ , and eventually  $\text{Co(OH)}_4^{2-}$ , often denoted as  $\alpha\text{-Co(OH)}_2$ . As the applied potential is increased,  $\alpha\text{-Co(OH)}_2$  would suffer a phase transition, transforming into  $\beta\text{-Co(OH)}_2$ . Then, the oxidation would occur in both  $\alpha\text{-Co(OH)}_2$  and  $\beta\text{-Co(OH)}_2$ , ultimately giving rise to  $\text{CoOOH}$ , which represented the authentic catalytic species involved in the OER.

Many oxide-based electrocatalysts for the OER undergo the formation of amorphous metal hydroxide layers during catalysis, which generally depends on the reaction conditions and material composition. The formation of amorphous surface layers commences at the surface and extends into the internal regions, suggesting that these amorphous layers exhibit porosity, enabling the diffusion of  $\text{H}_2\text{O}$  and its subsequent reaction with the bulk oxide.<sup>76</sup> For some metal (such as Fe, Co and Ni) oxides, the amorphous hydroxide layer is indeed produced through oxide hydration and exhibits a structural similarity to a layered double hydroxide (LDH).<sup>77</sup> The high oxidation potential exhibited by  $\text{Co}^{\text{III}}$  or  $\text{Fe}^{\text{III}}$  species within the amorphous layer is conducive to the dissociation of  $\text{H}_2\text{O}$ .<sup>78</sup> The dissolution and subsequent re-deposition of metal ions have been conspicuously observed in the realm of these composite oxide catalysts. In addition, perovskite materials can similarly undergo transformation into amorphous layers during catalysis, a phenomenon driven by element-specific dissolution mechanisms.<sup>79</sup> As for other metal compounds, such as phosphides, carbides, nitrides or sulfides, the generation of hydroxide layers is attributed to the oxidation of the original catalyst by oxygen-containing species, followed by the hydrolysis of the formed oxide product. In fact, these metal-based compounds can also serve as photocatalysts or cocatalysts in photocatalytic water splitting. Similar to the electrocatalytic reaction process, these metal-based photocatalysts also inevitably experience phase transformation. For example, the metallic Ni, employed as a cocatalyst for photocatalytic HER, could undergo progressive oxidation initiated by photogenerated holes, ultimately dissolving in  $\text{H}_2\text{O}$  and leading to the creation of a hollow NiO structure.<sup>80</sup> In the context of  $\text{NiO}_x$  species functioning as cocatalysts in the photocatalytic OER, these species could be oxidized to  $\text{NiO(OH)}$  when exposed to photogenerated holes, and these  $\text{Ni(III)}$  species in  $\text{NiO(OH)}$  were the essential active sites for the OER.<sup>67,80,81</sup>

### 2.2.3 Structural reconstruction

**2.2.3.1 Size.** A prevalent structural change observed in materials during catalysis is the alteration of the particle size. The sintering of catalysts, which leads to a reduction in the specific surface area and potential change in the coordination environment, is a key factor in irreversible deactivation. In the context of metal nanoparticle sintering on a support, two principal

mechanisms are proposed: the dissolution of metal into the electrolyte followed by its subsequent redeposition on the support, a phenomenon known as Ostwald ripening, or direct particle coalescence.<sup>82</sup> When a Pt catalyst is employed in a proton-exchange membrane fuel cell, the impetus driving Ostwald ripening and particle coalescence stems from a redox transformation involving the conversion between metallic  $\text{Pt}^0$  and  $\text{Pt}^{2+}/\text{Pt}^{4+}$ , which is markedly expedited during voltage cycling and intricately linked to the catalyst's morphological characteristics.<sup>83</sup> Specifically, Pt undergoes size-dependent oxidation to  $\text{Pt}^{2+}$  during electrocatalysis. Subsequently, the oxidized Pt species may dissolve in acidic electrolytes, and they can either migrate to the anode for reduction by  $\text{H}_2$ , resulting in the formation of large nanoparticles, or they may undergo re-reduction at the cathode.

Likewise, the sintering of a metal co-catalyst can take place when subjected to light irradiation and  $\text{H}_2\text{O}$  vapor. When compared to metal nanoparticles, subnanometer metal species, including isolated atoms or clusters, used as cocatalysts on semiconductors, exhibit enhanced photocatalytic performance.<sup>84,85</sup> However, Pt atoms and clusters, commonly used as cocatalysts in the photocatalytic HER, may sinter, yielding more active Pt nanoparticles ( $\sim 1$  nm) compared to isolated Pt atoms or larger counterparts.<sup>86,87</sup> During the photocatalytic water splitting, the photogenerated electrons and hydrogen-containing intermediates can induce the reduction of  $\text{Pt}^{2+}$  cations to highly dispersed Pt nanoparticles, and the size of partially oxidized Pt co-catalysts can be maintained at less than 2 nm by precise control, contributing to the efficient photocatalytic overall water splitting by effectively inhibiting the reverse reaction between  $\text{H}_2$  and  $\text{O}_2$ .<sup>75</sup> Due to their high associated surface free energy, transition metal oxide photocatalysts are prone to sintering when particle sizes are less than 10 nm. As a typical example, considering CoO nanoparticles with sizes below 10 nm, the sintering phenomenon observed during photocatalytic overall water splitting can be attributed to the generation of amorphous  $\text{Co(OH)}_2$  on the particle surface and the subsequent dehydrative coalescence.<sup>88</sup>

In addition, the dissolution and redeposition process of the catalyst within the electrolyte exerts a significant impact on its size. The removal of atoms from the catalyst surface can directly dissolve into the electrolyte, resulting in a reduction in the catalyst size. Nevertheless, dissociated metal atoms located near the cathode have the potential to redeposit on the catalyst surface, leading to the enlargement of existing particles or the formation of new clusters. These clusters and catalyst fragments can aggregate, forming agglomerations of varying sizes and shapes. Furthermore, the Ostwald ripening triggers the transportation of atoms from small to larger nanoparticles, and even facilitates substance redistribution in the process. Taking the complete restructuring of Cu nanoparticles as an example, the process involved three sequential stages: the prompt oxidation to  $\text{Cu}_2\text{O}$  upon exposure to the electrolyte, Cu ion dissolution during the open circuit potential state, the reintegration of aqueous Cu ions into the nanoparticle surface at the startup stage ( $0.1 V_{\text{RHE}}$ ). Ultimately, particle size redistribution occurs,



resulting in the disappearance of smaller particles during the operational stage (Fig. 5d).<sup>49</sup>

**2.2.3.2 Morphologies.** The migration of just a small number of atoms within each crystal can result in the exposure of different facets on the particles, often accompanied by morphological changes. Notably, metal nanocrystals demonstrate varying selectivity and activity profiles on their distinct facets. As exemplified in the case of metal Cu catalysts employed in the electrocatalytic CO<sub>2</sub>RR, Cu (100) facets exhibited a preference for generating C<sub>2+</sub> products, especially C<sub>2</sub>H<sub>4</sub>, whereas Cu (111) facets tended to favor CH<sub>4</sub> production. Compared to Cu (111) facets. The lower coupling barrier and the higher adsorption energy of \*CO endow Cu (100) facets with enhanced performance in facilitating C–C coupling through \*CO dimerization and subsequent hydrogenation into C<sub>2+</sub> products.<sup>89</sup> When metal atoms partake in interactions with metal–CO<sub>2</sub> and metal-intermediates, accompanied by electron transfer from metal to the bound CO<sub>2</sub> or intermediate, it is plausible that the metal atoms may undergo positional changes on the catalyst surface. This dynamic process could potentially lead to an overall morphological transformation of the catalyst. For instance, under the CO<sub>2</sub>RR reaction conditions, the degradation of Cu nanocubes was noted to coincide with the vanishing of the (100) facets, which was associated with a decrease in C<sub>2+</sub> selectivity since the (100) facets played a favorable role in promoting C–C coupling.<sup>90</sup>

In contrast to the evident morphological alterations observed in pure metallic catalysts, metal compounds undergo a transformation into amorphous oxide or oxyhydroxide layers, typically with a thickness ranging from 2 to 10 nm during the OER process. The discernibility of this change depends on the specific morphology of the precatalyst. As reported, the boronized metal sheets would undergo the dynamic evolution of metaborate-containing oxyhydroxide thin films (2–5 nm) during the OER process. Remarkably, these ultra-thin oxyhydroxide films exerted no significant impact on the overall morphology of the material. However, for mackinawite FeS nanosheets grown on iron foam with unique nanoarray structures, obvious morphology changes could be observed.<sup>91</sup> As shown in Fig. 5e, core@shell iron@iron oxysulfide nanoparticles as the catalytically active phase are generated *in situ* on FeS nanosheets during the HER testing, and FeS nanosheets totally transform into a porous amorphous FeO<sub>x</sub> film during the OER testing.<sup>50</sup> Strikingly, amorphous CoS<sub>x</sub> nanoboxes underwent a remarkable transformation into low-crystallinity CoOOH nanoplates due to the oxidizing conditions required for the OER<sup>51</sup> (Fig. 5f).

In this section, we comprehensively outline the dynamic transformations involving alterations in surface properties, composition, and structure. These profound changes inherently encompass subtle, low-level transformations. In other words, structural evolution is frequently concomitant with variations in components. As these components change, the chemical state and coordination environment generally undergo corresponding alterations. It is worth noting that some studies have reported the reversibility of dynamic

transformation on photo/electrocatalysts. As a typical example, the well-known MOF-74 demonstrated a “self-healing” phenomenon.<sup>26</sup> It initially transformed into Ni<sub>0.5</sub>Co<sub>0.5</sub>(OH)<sub>2</sub> and Ni<sub>0.5</sub>Co<sub>0.5</sub>OOH<sub>0.75</sub> species under different applied potentials and then reverted to its original structure after 15 days in ambient air. In addition, Ni<sub>3</sub>N, an efficient OER catalyst, exhibited an intermediate conversion from hydroxide/oxide to oxyhydroxide.<sup>92</sup> Importantly, the transformation from hydroxide/oxide to oxyhydroxide was reversible, contrasting with the irreversible nature of the transition between Ni<sub>3</sub>N and hydroxide/oxide. The complexity of photo/electrocatalysis introduces numerous variables affecting dynamic transformation reversibility. Variations in electrolytes and catalyst surfaces can introduce changes in dynamic evolution, even when considering the same material and reaction type. Therefore, reversibility in one case does not necessarily apply to an entire material class. Further research should focus on exploring the reversibility of photo/electrocatalysts.

### 2.3 Identification

The identification of the dynamic transformation of catalysts during the photo/electrocatalytic reaction can be achieved by a series of experiments and characterization studies. Over an extended past time frame, the dynamic transformation of photo/electrocatalysts before and after reactions has been primarily investigated through the utilization of *ex situ* characterization and electrochemical measurements. Nonetheless, these *ex situ* characterization studies frequently encounter challenges in delivering compelling results due to the scarcity of short-lived intermediates, the potential reversibility of reconstruction, and the inherent issues of catalyst aging, damage and contamination during the sample preparation and measurement procedures. Moreover, the indirect evidence presented by electrochemical measurements, including variations in oxidation peaks, changing trends in catalytic current density, and fluctuations in charge transfer resistance, may fall short of offering robust substantiation for catalyst reconstruction phenomena. With the rapid advancement of *in situ/operando* techniques, real-time information can be acquired to gain comprehensive insights into the intricate dynamic transformation mechanism of catalysts.<sup>93</sup> In the following section, we will provide a concise overview of the key attributes of frequently utilized *in situ/operando* techniques and expound upon their contributions to the elucidation of as-formed active species, the assessment of reversibility phenomena, and the exploration of catalytic mechanisms at the atomic level.

**2.3.1 *In situ/operando* X-ray characterization.** X-ray diffraction utilizes specific X-ray wavelengths to irradiate a sample, leading to the excitation of secondary X-rays of the atoms in the sample and the generation of distinctive diffraction patterns through their interference.<sup>94</sup> In the hard X-ray regime, X-ray absorption spectroscopy (XAS) plays a crucial role in discerning the local environment of active sites that govern reaction activity and selectivity.

In theory, the XAS spectra comprise two regions: X-ray absorption near edge structure (XANES), offering element-



specific electronic and bonding insights, and extended X-ray absorption fine structure (EXAFS), which unveils local structural characteristics, encompassing bond distances and coordination numbers surrounding the absorbing atoms. Recently, *operando* XAS spectra have been obtained to elucidate the sequential dynamic transformations from  $\text{CuN}_4$  to  $\text{CuN}_4/\text{Cu}$ -nanoclusters (NC), and subsequently evolving into  $\text{CuN}_3/\text{CuNC}$

during the electrocatalytic ORR process.<sup>95</sup> Concretely, the XANES spectra (Fig. 6a) exhibited an outstanding negative shift of the white-line peak edge with the applied potential, suggesting the gradual reduction of  $\text{Cu}^{2+}$  to  $\text{Cu}^+$  during the cathodic ORR process. Furthermore, the primary Cu–N bond at approximately 1.5 Å could be evidently observed in the EXAFS spectra of the Cu K-edge. Furthermore, additional fitting curves

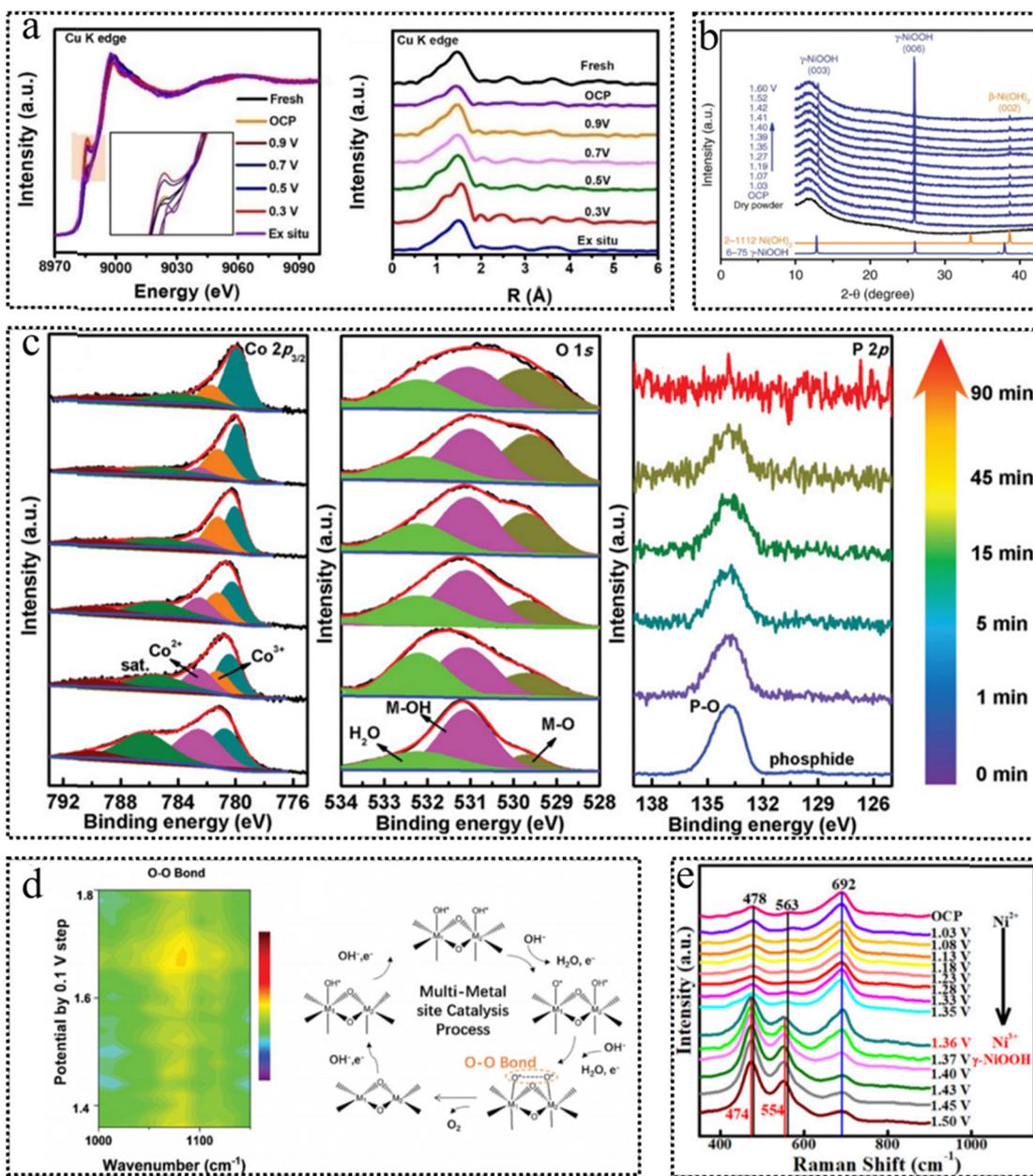


Fig. 6 (a) *Operando* Cu K-edge XANES spectra at different states and Cu K-edge FT-EXAFS spectra of Cu–N–C/GC. This figure has been reproduced from ref. 95 with permission from Wiley-VCH, copyright 2022. (b) *Operando* XRD patterns of  $\text{Ni(OH)}_2/\text{NF}$  at different potentials. This figure has been reproduced from ref. 96 with permission from Nature, copyright 2018. (c) *Operando* Co  $2p_{3/2}$ , O  $1s$  and P  $2p$  XPS spectra of  $\text{Co}_{0.8}\text{Fe}_{0.2}\text{P@C}$ . This figure has been reproduced from ref. 97 with permission from Wiley-VCH, copyright 2020. (d) 2D contour image of the *operando* ATR FT-IR spectrum for the OER (1.4–1.8 V vs. RHE) using 1 M KOH in  $\text{H}_2^{16}\text{O}$  and schematic illustration of the multi-metal-site catalytic mechanism. This figure has been reproduced from ref. 98 with permission from Royal Society of Chemistry, copyright 2020. (e) *Operando* Raman spectra of  $\text{NFO}/\text{NF}$  with photothermal effect (*i.e.*, irradiated with 808-nm NIR light) from OCP to 1.50 V vs. RHE in 1.0 M KOH. This figure has been reproduced from ref. 99 with permission from American Chemical Society, copyright 2021.



illustrated a reduction in the coordination number of the Cu–N bond from 4.5 to around 2.8 during the ORR within the 0.9–0.3V potential range. Simultaneously, Cu–Cu bonds with a coordination number of approximately 3 emerged, suggesting the generation of Cu NC. In the realm of *in situ/operando* investigations, diffraction patterns serve as invaluable tools for analyzing the dynamic alterations in the phase structure and crystal size of bulk photo/electrocatalysts. As an illustrative example, *in situ* XRD patterns (Fig. 6b) of Ni(OH)<sub>2</sub> provided compelling evidence of the presence of  $\gamma$ -NiOOH at distinct electrochemical potentials, as manifested by the conspicuous appearance of robust peaks.<sup>96</sup> Despite the benefits of rapid detection and ease of operation offered by XRD techniques, they fall short in accurately discerning the dynamic transformation of surface-reconstructed species with an amorphous state. X-ray photoelectron spectroscopy (XPS) is a surface-sensitive, quantitative technique known for its ability to provide precise, element-specific insights into electronic states and chemical composition by precisely measuring the kinetic energy of photoelectrons. As illustrated by the *operando* XPS spectra in Fig. 6c, the XPS peak intensities corresponding to Co<sup>3+</sup> and M–O of Co<sub>0.8</sub>Fe<sub>0.2</sub>P@C exhibited a noticeable increase, while a decline could be observed in the peak assigned to the P–O bond during electrochemical activation.<sup>97</sup> This result substantiated the emergence of active CoOOH species, concomitant with the leaching of P. It is noteworthy that due to its sensitivity to catalyst surface properties, XPS is susceptible to the influence of surface impurities.

**2.3.2 Spectroscopic characterization.** Fourier-transform infrared (FTIR) spectroscopy can furnish molecular absorption spectra that encompass information regarding the composition and structural characteristics of functional groups within newly formed species during the dynamic transformation. For example, the *operando* attenuated total reflection FT-IR (ATR FT-IR) spectrum shown in Fig. 6d provided direct evidence of O–O bond formation at 1078 cm<sup>-1</sup> in the potential range from 1.4 to 1.8 V vs. RHE.<sup>98</sup> With the potential increasing to 1.68 V vs. RHE, there was a notable rapid increase in peak intensity, which indicated the substantial generation of a multitude of intermediate species during the OER process, resulting from the formation of metal oxyhydroxides. However, it is important to note that FTIR spectroscopy has a relatively lower sensitivity, which necessitates a high content of detected components in samples for accurate analysis.

Raman spectroscopy, when coupled with probing molecules, provides a means to access surface information about catalysts during catalysis. This technique exhibits a high sensitivity for monitoring the transformation of chemical bonds and functional groups within the bulk catalysts. Note that Raman spectroscopy can identify metal-related bonds and key intermediate species through low-wavenumber signals. The *operando* Raman measurements (Fig. 6e) were performed to detect the dynamic transformation of NiFe<sub>2</sub>O<sub>4</sub> during the electrocatalytic OER process.<sup>99</sup> The active vibrational modes spanning 400 to 700 cm<sup>-1</sup> demonstrated potential-dependent behavior, suggesting dynamic phase transformations occurring at the

electrode surface within the potential range from open circuit potential (OCP) to 1.50 V. Upon reaching a potential of 1.41 V, the appearance of peaks at 474 and 554 cm<sup>-1</sup>, which correspond to  $\gamma$ -NiOOH, indicates the generation of Ni<sup>3+</sup> species on the electrode surface.

**2.3.3 Microscopic characterization.** The structural changes and phase transformation of the catalysts during the reaction process can be monitored *via operando* microscopic characterization, encompassing (scanning) transmission electron microscopy [(S)TEM] and identical location TEM (IL-TEM) and atomic force microscopy (AFM). Operando (S)TEM provides a unique capability for real-time visualization of morphological and compositional changes of pre-catalysts by employing precisely engineered operational holders and reaction cells. For example, the progressive evolution of (001)-oriented CuO nanosheets into highly branched metallic Cu<sup>0</sup> dendrites was captured through *in situ* TEM.<sup>100</sup> As depicted in Fig. 7a, the facet-oriented CuO nanosheets underwent a swift transformation into nanoscale reactive sheet fragments, which were then slowly reconstituted into stable, larger agglomerates and dendritic Cu structures. The transformation rate was contingent upon the applied potential or the duration of the reaction.

IL-TEM offers the advantage of monitoring the transformation of identical particles before and after the reaction, enabling precise comparative analysis. IL-TEM was conducted to monitor the morphological evolution of the Ba<sub>0.5</sub>Sr<sub>0.5</sub>Co<sub>0.8</sub>Fe<sub>0.2</sub>O<sub>3- $\delta$</sub>  (BSCF) perovskite during the OER process. Although the IL-TEM images displayed an unaltered morphology of a BSCF particle, the high-resolution identical location high-angle annular dark-field STEM (IL-HAADF-STEM) images (Fig. 7b) exhibited the propensity of the BSCF surface to become a porous configuration.<sup>101</sup> Notably, this porosity was observed to increase with an increase in the number of cycles. Due to the existence of a high-energy electron beam, TEM imaging has the potential to induce changes or even irreversible damage to the structure of low-crystalline catalysts. This raises the concern that the observed structural evolution during *in situ* experiments may be attributed to electron beam-induced irradiation effects.

AFM has emerged as a widely adopted non-contact technique for surface characterization in contemporary research. A tip with atomic sharpness scans across the surface, systematically probing variations in the interatomic forces between the tip and the surface atoms. Subsequently, the surface's atomic-scale topography is meticulously mapped by recording the positional adjustments of scanning points corresponding to the micro-cantilever. As depicted in Fig. 7c, the *operando* electrochemical AFM measurements provided clear evidence of the surface roughening of the Cu nanocubes, the vanishing of (100) facets, the emergence of pores, the depletion of Cu, and the reduction of CuO<sub>x</sub> species during the electrocatalytic CO<sub>2</sub>RR.<sup>102</sup>

The capabilities, advantages and disadvantages of the above-mentioned *in situ/operando* characterization techniques for identifying dynamic remodeling are summarized in Table 1 and Fig. 8. For the identification of dynamic reconstructions in surface chemical states and coordination environments, XAS,



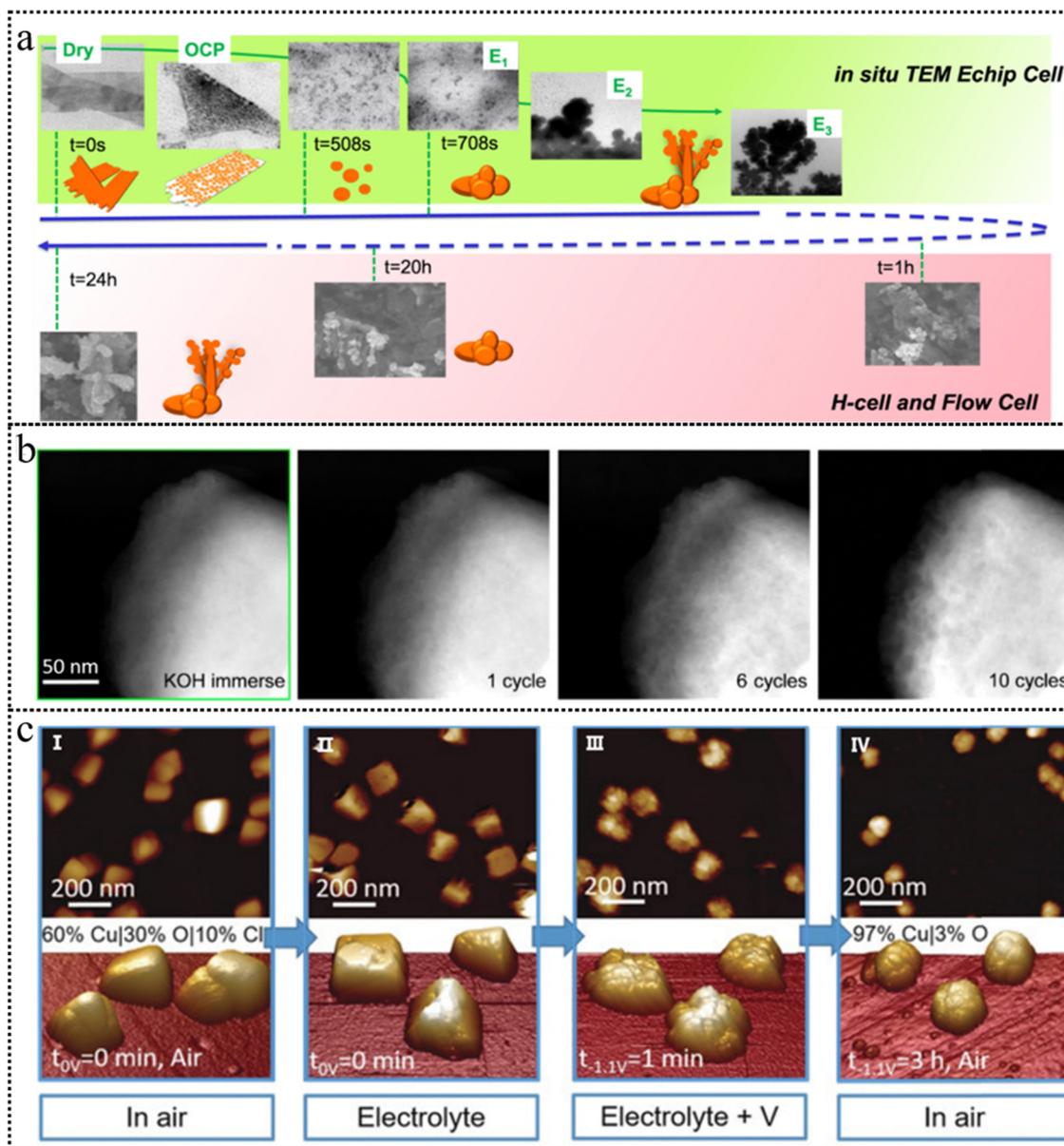


Fig. 7 (a) A schematic timeline illustrating the experimentally observed evolution of the CuO NS morphology probed through an *in situ* TEM E-chip flow cell, H-cell, and flow cell electrolyzer. This figure has been reproduced from ref. 100 with permission from Nature, copyright 2021. (b) HAADF-STEM images of the BSCF catalyst after cyclic voltammetry, conducted within a potential range of 1.1–1.8 V vs. RHE at a scan rate of 10 mV s<sup>-1</sup>. This figure has been reproduced from ref. 101 with permission from American Chemical Society, copyright 2020. (c) AFM images of Cu cubes electrodeposited on highly oriented pyrolytic graphite (HOPG) acquired in air (I), and *operando* EC-AFM measurements in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> aqueous solution at open circuit potential (II), at -1.1 V versus RHE in the same electrolyte for 1 minute (III) and after 3h under the same conditions as in (III) and subsequent air exposure (IV). This figure has been reproduced from ref. 102 with permission from Wiley-VCH, copyright 2018.

XPS, Raman and FT-IR spectroscopy are ideally suited due to their unique capabilities. XRD is instrumental in monitoring phase transformations and variations in crystal size during catalytic reactions. Larger-scale dynamic changes, such as those in size and morphology, can be observed using SEM, TEM and AFM. Additionally, SEM and TEM equipped with Energy Dispersive Spectroscopy (EDS) are useful for detecting elemental redistribution, while high-resolution TEM also excels in observing phase transformations. Theoretically, only the integration of multiple *in situ/operando* characterization techniques can

furnish substantial evidence for the dynamic transformation occurring in active sites. Future research should place an emphasis on resolving the impact of instrumentation and operating conditions on catalysts during *in situ/operando* characterization.

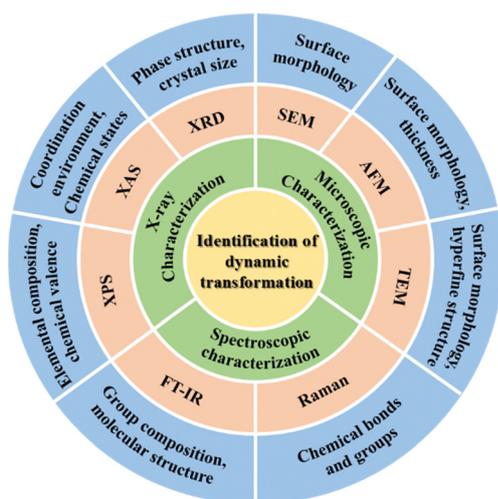
### 3. Triggers of reconstruction

Photo/electrocatalysis relies on catalysts, electrolytes, and applied bias as primary components. Catalysts possess intrinsic



Table 1 The main *in situ/operando* characterization for identifying the dynamic reconstruction of photo/electrocatalysts

Types	Techniques	Capabilities	Advantages	Disadvantages
X-ray characterization	XAS	Chemical states and coordination environments	High precision and extensive applicability	Vulnerable to mutual elemental interference and overlapping peaks
	XRD	Phase structure and crystal size	Rapid detection and straightforward operation	Not suitable for amorphous structure testing
	XPS	Elemental composition and chemical valence states	High sensitivity and excellent spectral repeatability	Fail to detect the bulk phase and sensitive to surface impurities
Spectroscopic characterization	FT-IR	Group composition and molecular structure	High penetration	Limited sensitivity
	Raman	Chemical bonds and groups	High sensitivity with rapid testing	Sensitive to optical parameters
Microscopic characterization	SEM	Surface morphology	Stereoscopic and wide-field imaging	Low resolution and inability to show fine structure
	AFM	Surface morphology and thickness	Stereoscopic and wide-field imaging	Low resolution and inability to show fine structure
	TEM	Surface morphology and hyperfine structure	High resolution and hyperfine structure clarity	Only captures two-dimensional information

Fig. 8 *In situ/operando* characterization techniques for identifying internal electric fields and their specific functions.

physicochemical attributes that govern their self-construction. This section delves into the myriad factors influencing this self-assembly to enhance overall.

### 3.1 Single-atom catalysts

Single-atom catalysts, including cluster catalysts, have been intensively studied in catalysis due to their exceptional atomic efficiencies and heightened activities.<sup>103–105</sup> However, prevailing research typically adheres to the synthesis–characterization–performance testing paradigm. Consequently, our understanding of catalyst active sites predominantly adopts a “static” perspective, disregarding the dynamic structural evolution of single atoms amidst catalyst activation and reaction conditions. This oversight poses critical challenges. On the one hand, single-atom catalysts may undergo reconstruction during catalytic reactions, creating disparities between the actual active sites and the initial catalytic configurations.<sup>106</sup> Conversely, uncontrolled active site reconstruction might hasten catalyst degradation. Hence, leveraging the interaction between

single atoms and their support becomes pivotal in regulating the extent of reconstruction.

Small metal species like single atoms and nanoclusters possess high surface energy, predisposing them to aggregate, especially under reductive potential. This tendency often proves detrimental, leading to catalytic performance decline.<sup>107–110</sup> Paradoxically, such aggregation might also facilitate catalytic reactions by transforming single atoms into nanoclusters, thereby assuming the role of actual active sites. Wei *et al.* reported a novel catalytic urea synthesis method, using Cu single atoms incorporated on a ceria support as the catalyst, to directly couple nitrate and carbon dioxide.<sup>111</sup> The paper reveals the structural evolution of the catalyst during urea synthesis reaction by using XAS and IR spectroscopy techniques. From the *operando* XAS results depicted in Fig. 9a and b, it was found that Cu single atoms reconstitute into copper clusters at reducing potentials, and the copper clusters are the real active sites. Additionally, the schematic diagram in Fig. 9c illustrates the dynamic and reversible transformation of Cu<sub>4</sub> into Cu<sub>1</sub> configurations when switching the applied potential to the open-circuit potential. The paper also investigates the reaction mechanism of the catalyst by using density functional theory calculations and kinetic measurements and finds that copper clusters can effectively adsorb and activate nitrate and carbon dioxide, and significantly reduce the activation energy of O–O coupling. Another work done by Dilan Karapinar *et al.* shows the same phenomenon of Cu single atoms catalysts.<sup>112</sup> *Operando* XAS analysis reveals that the isolated sites are gradually transformed into metallic copper nanoparticles under working conditions. These metallic copper nanoparticles are believed to be the catalytically active species. Interestingly, from the XAS characterization of post-electrolysis samples without applying voltage, the disappearance of the metallic copper phase and the restoration of the original spectra illustrate that the restructuring behavior is reversible (Table 2).

Yang *et al.* used *operando* XAS combined with transmission electron microscopy (TEM) to reveal that the synthesized Cu–N<sub>4</sub> single-atom site underwent restructuring, forming



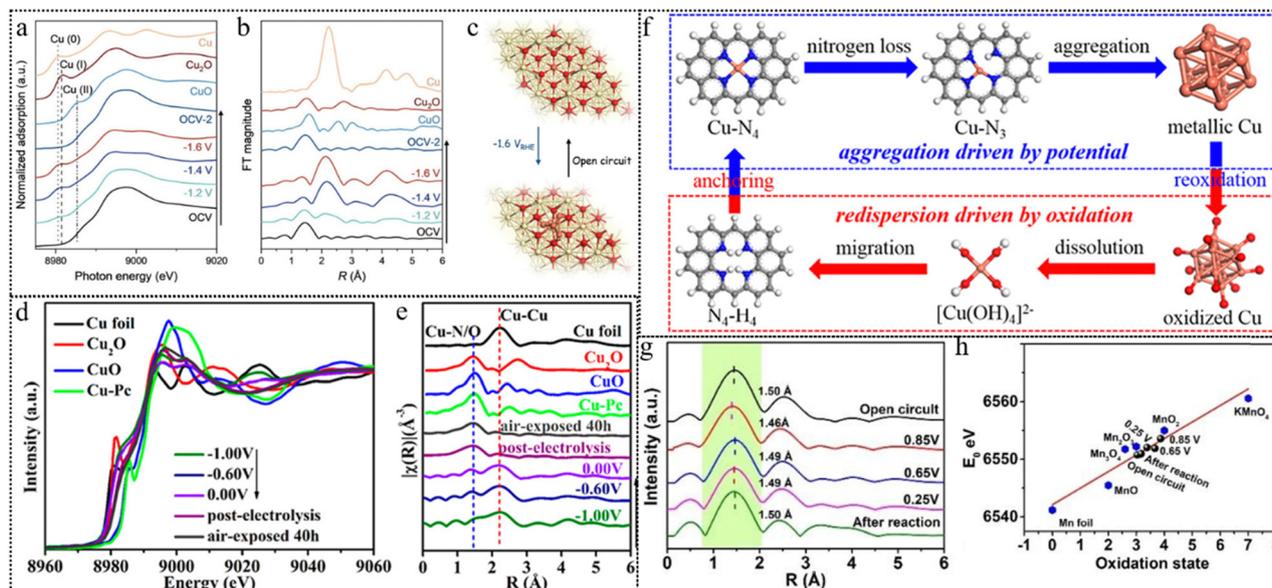


Fig. 9 (a) Cu K-edge XANES spectra of  $\text{Cu}_1\text{-CeO}_2$  recorded at different cathodic potentials during C–N coupling. (b) Cu R-space EXAFS spectra of  $\text{Cu}_1\text{-CeO}_2$  recorded at different cathodic potentials during C–N coupling. (c) Schematic diagram of reconstitution of copper single-atoms to clusters suggested by the *operando* XAS measurements. This figure has been reproduced from ref. 111 with permission from Wiley-VCH, copyright 2023. (d) Cu K-edge XANES spectra at different potentials from  $-1.00$  to  $0.00$  V vs. RHE. (e) Cu K edge FT-EXAFS spectra by switching potentials from  $-1.00$  to  $0.00$  V vs. RHE. (f) Proposed mechanisms of aggregation driven by the applied potential and re-dispersion in the oxidation environment. This figure has been reproduced from ref. 113 with permission from American Chemical Society, copyright 2022. (g) Mn K-edge FT-EXAFS spectra at open circuit, different applied potentials, and after reaction. (h) Average oxidation state of Mn in the Mn single atom sites at open circuit,  $0.85$  V,  $0.65$  V,  $0.25$  V and after reaction compared with Mn foil, MnO,  $\text{Mn}_2\text{O}_3$ ,  $\text{Mn}_3\text{O}_4$ ,  $\text{MnO}_2$  and  $\text{KMnO}_4$ . This figure has been reproduced from ref. 114 with permission from Wiley-VCH, copyright 2023.

nanoparticles of about  $5$  nm during the chemical process of nitrate reduction to ammonia.<sup>113</sup> According to the *operando* XAS measurements in Fig. 9d and e, the reduction of  $\text{Cu}^{2+}$  to  $\text{Cu}^+$  and  $\text{Cu}^0$ , followed by the aggregation of copper single atoms, aligned with the increase in the  $\text{NH}_3$  production rate induced by the applied potential, as the applied potential scanning from  $0.00$  V to  $-1.00$  V. At  $-1.00$  V, the catalyst achieved its peak  $\text{NH}_3$  production rate at  $4.5$   $\text{mg cm}^{-2} \text{h}^{-1}$  ( $12.5$   $\text{mol NH}_3 \text{ gCu}^{-1} \text{ h}^{-1}$ ), exhibiting a Faraday efficiency of  $84.7\%$ , surpassing the performance of most previously reported Cu catalysts. The proposed mechanism is depicted in Fig. 9f. When the applied potential was removed, the aggregated Cu nanoparticles (NPs) demonstrated a reversible transformation into single atoms, which subsequently reverted to the  $\text{Cu-N}_4$  structure upon exposure to the ambient atmosphere. This phenomenon typically obscures the potential-driven restructuring process that is challenging to observe during the reaction. These findings suggest that Cu NPs serve as the primary catalytic site for the reduction of nitrate to ammonia, corroborating the results obtained from post-deposited Cu NP catalysts and DFT calculations.

A single metal atom may also migrate instead of aggregation from one type of bonding site to another. Here is an example. The authors employ *operando* XAS to monitor the dynamic process of the Mn– $\text{N}_4$  configuration and the valence state of Mn atoms throughout the oxygen reduction reaction (ORR).<sup>114</sup> Through analysis of the Mn K-edge FT-EXAFS spectra in Fig. 9g,

it is observed that the primary scattering peak, indicative of the Mn–N/C bond length, shifts successively from  $1.50$  to  $1.46$  Å, and then to  $1.49$  Å. This shift illustrates the transformation of the Mn– $\text{N}_4$  configuration into Mn– $\text{N}_3\text{C}$  and subsequently into Mn– $\text{N}_2\text{C}$  under varying applied potentials. The linear fitting results according to the XANES spectra are presented in Fig. 9h. During the transformation of the coordination environment of Mn, the valence state of Mn changes from  $+3.0$  to  $+3.8$  and then to  $+3.2$ .

In addition to the involvement of single atom migration in the reconstruction, single-atom modification can also serve to anchor active sites, thereby preventing excessive reconstruction. For instance, Sun *et al.* demonstrated that a catalyst with Ru atoms atomically dispersed on a nickel–vanadium ( $\text{Ni}_3\text{V}$ ) layered double hydroxide (LDH) support, achieved through a robust atomic metal–support interaction, exhibited remarkable activity in both the HER and OER.<sup>115</sup> *In situ* XAS and *operando* Raman spectroscopy studies have shown that the presence of Ru atoms on the surface of nickel–vanadium LDH plays a crucial role in stabilizing the surface rich in dangling bonds, further leading to a non-reconstructed surface. Due to the robust metal–support interaction on nickel–vanadium LDH, the substantial interaction effectively increased the stability of the reactive atomic Ru sites, minimizing oxidation state fluctuations during the cathodic HER without necessitating reconstruction. In addition, Ru atomic sites can stabilize Ni sites, rendering them more resilient to bond contraction and





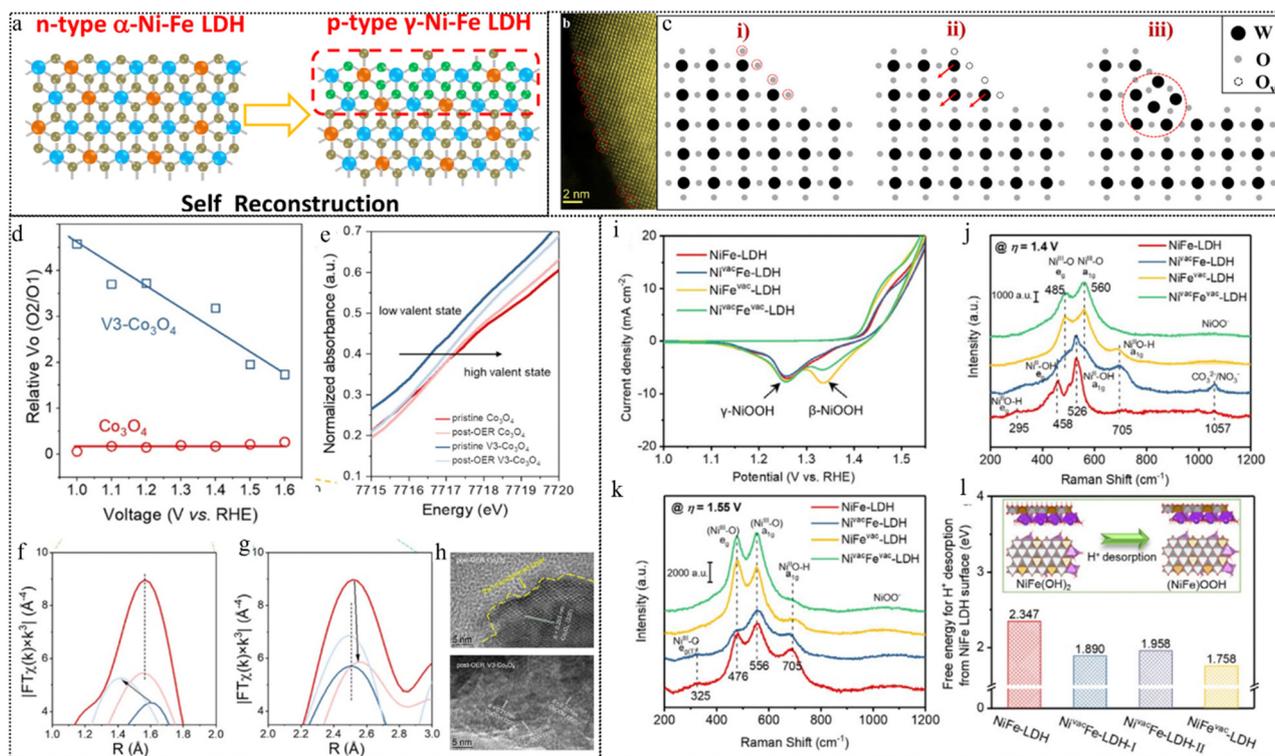
**Table 2** Summary of intrinsic triggers driving photo/electrocatalyst reconstruction and corresponding catalyst characterization

Intrinsic triggers	Reconstruction types	Precatalysts	Reconstructed species	Reaction/electrolyte	Characterization	Ref.	
Single-atom catalysts	Surface properties	Cu <sub>1</sub> -CeO <sub>2</sub>	Cu <sub>4</sub> -CeO <sub>2</sub>	Urea synthesis/0.1 M KHCO <sub>3</sub> with 50 mM KNO <sub>3</sub>	TEM, STEM, and <i>Operando</i> XAS	111	
	Surface properties	Cu-N <sub>4</sub>	Metallic Cu nanoparticles	CO <sub>2</sub> RR/0.1 M CsHCO <sub>3</sub>	TEM, STEM, and <i>Operando</i> XAS	112	
	Surface properties	Cu-N <sub>4</sub>	Metallic Cu nanoparticles	Reduction of nitrate to ammonia/0.1 M KOH including 0.1 M KNO <sub>3</sub>	TEM, STEM, and <i>Operando</i> XAS	113	
	Composition changes	Ru/Ni <sub>3</sub> V-LDH	Ru-O-Ni	OER&HER/1.0 M KOH	TEM, STEM, and <i>Operando</i> XAS/Raman	115	
Defects engineering	Surface properties	Mn-N <sub>4</sub>	Mn-N <sub>3</sub> /C, Mn-N <sub>2</sub> /C, and Mn-N/C	ORR/0.1 M KOH	TEM, STEM, and <i>Operando</i> XAS	114	
	Composition changes	n-type α-Ni-Fe LDH	p-type γ-Ni-Fe LDH	OER/1.0 M KOH	TEM, STEM, XAS, EPR, and FTIR	116	
	Surface properties	(110) facet of WO <sub>3</sub>	5-fold symmetry pentagonal column structure	—	Cs-STEM	117	
	Composition changes	Co <sub>3</sub> O <sub>4</sub>	Co(OH) <sub>2</sub> intermediate with proton vacancies (Co(II)O <sub>x</sub> (OH) <sub>y</sub> )	OER/1.0 M NaOH	TEM, STEM, XAS, and <i>Operando</i> Raman/ATR-IR/TEM	118	
	Composition changes	NiFe LDH	γ-(NiFe)OOH	OER/1.0 M KOH	TEM, STEM, XAS, and Raman	119	
	Composition changes	VNi-α-Ni(OH) <sub>2-x</sub>	NiOOH	OER/1.0 M KOH, UOR/1.0 M KOH + 0.33M urea	TEM, STEM, XAS, ESR, and PAS	120	
	Composition changes	CoF <sub>2</sub> porous nanowires	F-CoOOH nanosheets	OER/1.0 M KOH	TEM, STEM, and XAS	121	
	Structural reconstruction						
	Composition changes	Ni <sub>1-x</sub> Fe <sub>x</sub> PS	Disordered MOOH@MxSO <sub>y</sub> @Mx-PO <sub>y</sub> (M = Ni, Fe)	OER/1.0 M KOH	TEM, XAS, and Raman	122	
	Composition changes	Mo/N substituted CoO	CoOOH	OER/1.0 M KOH	Cs-STEM/TEM, real-time XPS	123	
	Composition changes	Pr <sub>3</sub> Ir <sub>1-x</sub> Mo <sub>x</sub> O <sub>7</sub>	Ir-Obri-Mo species	OER/0.1 M HClO <sub>4</sub>	TEM, EPR	124	
	Heterostructure construction	Surface properties	NiSe <sub>2</sub> /FeSe <sub>2</sub>	N(III)-O-Fe(III)	OER/1.0 M KOH	TEM, XAS, and Raman	125
		Composition changes	LaNiO <sub>3</sub> /NiO	LaNiO <sub>3</sub> /NiOOH	OER/1.0 M KOH	TEM, STEM, XAS, and <i>Operando</i> Raman	126
		Composition changes	Core/shell Pd/Fe <sub>3</sub> O <sub>4</sub> nanoparticles	Core/shell Pd/Fe <sub>x</sub> (OH) <sub>2-2x</sub> nanoparticles	HER/0.1 M KOH	TEM, XAS	127
Composition changes		Ni <sub>1-x</sub> P <sub>x</sub> /MoO <sub>x</sub>	Ni-P-O	OER/1.0 M KOH	TEM, Raman, FTIR, and	128	
Composition changes		LaNiO <sub>3</sub>	VLa-LaNiO <sub>3</sub> /NiOOH	OER/1.0 M KOH + 0.1 M H <sub>2</sub> SO <sub>4</sub> /0.1 M H <sub>2</sub> CO <sub>3</sub> /0.1 M HNO <sub>3</sub>	TEM	129	
Structural reconstruction							
Composition changes		Spinel Li <sub>2</sub> Co <sub>2</sub> O <sub>4</sub>	Amorphous Li <sub>2-x</sub> Co <sub>2</sub> O <sub>4-δ</sub> (OH) <sub>δ</sub> (Co <sup>3+</sup> /Co <sup>4+</sup> )	OER/1.0 M KOH	TEM, STEM, XAS, and Raman	130	
Structural reconstruction							
Surface properties		Co <sub>2</sub> (OH) <sub>3</sub> Cl	β-CoOOH	OER/1.0 M KOH/1.0 M PBS buffer	TEM, STEM, and XAS	131	
Composition changes							

Table 2 (continued)

Intrinsic triggers	Reconstruction types	Precatalysts	Reconstructed species	Reaction/electrolyte	Characterization	Ref.
	Composition changes	IrFeCoNiCu HEA	Ir rich shell	OER/0.1 M HClO <sub>4</sub>	TEM and STEM	132
	Composition changes	Fe <sub>0.01</sub> -Ni/Ni <sub>0.2</sub> Mo <sub>0.8</sub> N	Fe <sub>0.01</sub> /Mo-NiO	HER/1.0 M KOH/1.0 M KOH + natural seawater	TEM and <i>Operando</i> Raman	133
	Composition changes	Co-MOF/Fe <sub>10</sub>	CoFe LDH	OER/1.0 M KOH	TEM, STEM, and Raman	134
	Composition changes	NiFe PBA	NiOOH/FeOOH	OER/1.0 M KOH	TEM, STEM, Raman, and FTIR	135
	Composition changes	NiFeP	NiOOH/FeOOH	OER&HER/1.0 M KOH	TEM	136
	Composition changes	NiFe/NiS <sub>x</sub>	NiFe-LDH intercalated SO <sub>4</sub> and CO <sub>3</sub> anions	OER/1.0 M KOH	Raman and TOF-SIMS	33
	Surface properties	NiOOH	Fe <sub>x</sub> NiOOH	OER/1.0 M KOH + 10 μM Fe <sup>3+</sup>	XAS and XFM	137
	Composition changes	NiO	NiOOH/NiO and NiFe LDH/NiO	0.5 M borate buffer + 1 mM Fe(NO <sub>3</sub> ) <sub>3</sub>	EC-TEM, STEM, <i>Operando</i> XAS/Raman, and time-resolved ICP-MS	138
	Structural reconstruction	Bi-MOF	Bi <sub>2</sub> O <sub>2</sub> CO <sub>3</sub> and metallic Bi	0.1 M-1.0 M KHCO <sub>3</sub> aqueous solution	TEM, STEM, FTIR, and <i>Operando</i> Raman/ATR-SEIRAS	139
Crystallinity	Structural reconstruction	Crystalline Ni <sub>3</sub> Fe	Amorphous Ni <sub>3</sub> Fe	1 M ultra pure KOH	TEM, STEM, and <i>Operando</i> XAS	140
	Composition changes	Amorphous NiFeMo	NiFeOOH	OER/0.1 M KOH	TEM, STEM, XAS, UPS, and <i>Operando</i> Raman	141
	Structural reconstruction	Amorphous SrCo <sub>0.85</sub> Fe <sub>0.1</sub> P <sub>0.05</sub> O <sub>3-δ</sub>	Reconstructed SCFP	OER&HER/1.0 M KOH	TEM and XAS	142
	Composition changes	Amorphous Pt nanospheres	Crystalline clusters with Pt-O ligand	HER/0.5 M H <sub>2</sub> SO <sub>4</sub> , HOR/0.1 M HClO <sub>4</sub>	TEM, Raman, FTIR, and UV-vis	143
	Surface properties					
	Structural reconstruction	H-Co(OH) <sub>2</sub> @Fe(OOH) <sub>1-x</sub> Cl <sub>x</sub>	CoOOH@FeOOH	OER/1.0 M KOH	TEM, <sup>57</sup> Fe isotope labeled ICP-MS	144
	Composition changes					
	Surface properties					





**Fig. 10** (a) Schematic diagram of a p–n interface via the reconstruction of p-type  $\gamma$ -Ni–Fe LDH on n-type  $\alpha$ -Ni–Fe LDH. This figure has been reproduced from ref. 116 with permission from American Chemical Society, copyright 2021. (b) Atomic-resolution STEM-HAADF images of the surface atomic structure of  $\text{WO}_3$  with pentagonal columns. (c) The schematic images of surface reconstruction. This figure has been reproduced from ref. 117 with permission from American Chemical Society, copyright 2021. (d) Relative surface  $\text{V}_o$  ( $\text{O}_2/\text{O}_1$ ) contents of  $\text{Co}_3\text{O}_4$  and  $\text{V}_3\text{-Co}_3\text{O}_4$  in the *ex situ* XPS measurements under different applied potentials. (e) XANES spectra of  $\text{Co}_3\text{O}_4$  and  $\text{V}_3\text{-Co}_3\text{O}_4$  before and after the OER. (f) and (g) EXAFS spectra of  $\text{Co}_3\text{O}_4$  and  $\text{V}_3\text{-Co}_3\text{O}_4$  before and after the OER. (h) Post-OER HRTEM images of  $\text{Co}_3\text{O}_4$  and  $\text{V}_3\text{-Co}_3\text{O}_4$ . This figure has been reproduced from ref. 118 with permission from American Chemical Society, copyright 2023. (i) ECSA-normalized CV curves at a scan rate of  $10 \text{ mV s}^{-1}$ . (j), (k) Raman spectra at applied potentials of (j) 1.4 V and (k) 1.55 V vs. RHE. (l) DFT calculated free energy of  $\text{H}^+$  desorption from the surface of different NiFe-LDH-based structural models. This figure has been reproduced from ref. 119 with permission from Wiley-VCH, copyright 2021.

structural distortion induced by oxidizing Ni sites during the anodic potential. This stabilization facilitates an increase in the oxidation state of Ni sites, thereby enhancing the catalyst's OER performance. Unlike other bifunctional catalysts that necessitate structural reconstruction or transformation to accommodate HER/OER cycles, the Ru/Ni3V LDH catalyst possesses stable dual reaction sites and exhibits a robust metal–support interaction (*i.e.*, between Ru and Ni sites) during the water decomposition process.

### 3.2 Defect engineering

The deliberate introduction of defects significantly influences a catalyst's electronic configuration and the surrounding coordination environment of metal sites. This strategic maneuver accelerates photon/electron transfer, thereby augmenting catalytic performance.<sup>145,146</sup> Defect introduction stands as an effective means to regulate the reconstruction of catalysts. For example, thermodynamically equilibrium defects such as point defects (vacancies and interstitials) exhibit mobility or reactivity under external stimuli, such as light or specific potentials during catalysis.<sup>22,147–149</sup> This phenomenon promotes the formation of catalytic intermediates, rendering the system more stable through reconstruction. Conversely, line defects, though

non-equilibrium defects themselves, often undergo spontaneous reconstruction. However, their exploration within the catalysis field remains limited and lies beyond the scope of this discussion. Surface defects, such as grain boundaries, represent another common avenue for modulating catalyst performance. By impeding atomic migration, these boundaries offer a means for controlled reconstruction, thereby allowing for the adjustment of catalyst performance. These defects play a pivotal role in modulating the adsorption energies of active species of the catalyst's surface, thereby selectively stabilizing specific active species and fine-tuning catalytic performance.<sup>150,151,152</sup>

Inherent oxygen vacancies ( $\text{V}_o$ ) can facilitate the initiation of reconstruction during catalytic reactions by enhancing the photon/electron density around metal atoms and reducing the valence states of metal ions. Zhou *et al.* presented a novel strategy to synthesize ultrathin Ni–Fe LDH nanosheets with inherent  $\text{V}_o$  by a coprecipitation method. They then examined the influence of these vacancies on the catalyst structure and the performance in catalytic OER performance.<sup>116</sup> The study revealed that oxygen vacancies facilitate the surface reconstruction of Ni–Fe LDH, resulting in the generation of a p–n interface within the  $\gamma$ -Ni–Fe LDH/ $\alpha$ -Ni–Fe LDH heterostructure as shown in the schematic diagram in Fig. 10a, which can



enhance the electron density, reduce the oxidation state, optimize the OER reaction pathway, and lower the overpotential.  $V_O$  on the surface can induce surface structure reconstruction as a result of an enhancement in surface catalytic activity. Using spherical aberration-corrected scanning transmission electron microscopy (Cs-STEM) with atomic-scale precision, Meng *et al.* directly observed the surface restructuring on the step areas or high-index facets of  $WO_3$ .<sup>117</sup> The area with a significant abundance of  $V_O$  promotes the formation of a unique structure with 5-fold symmetry illustrated using the Cs-STEM image in Fig. 10b. As Fig. 10c shows the proposed reconstruction pathway, the presence of oxygen vacancies at the edge breaks the local symmetry and lowers the coordination number of W atoms outside. Consequently, these unsaturated coordinated W atoms tend to migrate into interstitial places to lower the surface energy finally forming such special structures. DFT calculations reveal that the step area requires much lower formation energy of oxygen vacancies which follows previous observation. This work presents that enhancement of surface reconstruction can be spontaneously achieved by high concentrations of oxygen vacancies. The introduction of oxygen vacancies often accompanies the structure reconstruction of the catalyst and the improvement of OER activity. As a skeleton mechanism of the OER, the LOM shows that the evolution of the  $O_2$  molecule involving lattice oxygen would generate oxygen vacancies that are occupied by the adsorbed  $OH^-$  in the electrolyte in the later cycles.<sup>150</sup> It means oxygen vacancies can affect and participate in structural reconstruction from the reaction mechanism.

Zhou *et al.* investigated the impact of  $V_O$  on structural reconstructing under OER conditions using a  $V_O$ -rich spinel  $Co_3O_4$  as a model catalyst.<sup>118</sup> From their XPS results (Fig. 10d), they observed that the comparative surface  $V_O$  concentration (ratio of  $O_2/O_1$  deconvoluted from XPS of O 1s) of  $Co_3O_4$  remains nearly constant regardless of the working potential, suggesting the inactive nature of the inherent low concentration of  $V_O$  and their limited transformation. Conversely, the relative  $V_O$  content of  $V_3-Co_3O_4$  steadily decreases with increasing positive potential, indicating a significant transformation of the introduced  $V_O$  during activation and the OER processes. Additionally, changes in the valence state of Co are confirmed by XAS of the near-edge structure (Fig. 10e). Analysis of the EXAFS spectral results (Fig. 10f and g) reveals that the Co–O distances for  $Co_3O_4$  exhibit a minimal variation after the OER, indicating the relatively stable spinel structure. In contrast, the Co–O distances decrease for  $V_3-Co_3O_4$  after the OER due to the formation of  $CoOOH$ . While disordered lattice fringes are observed only at the surface within a thin layer, indicating a low degree of surface reconstruction in  $Co_3O_4$ ,  $V_3-Co_3O_4$  exhibits a highly disordered lattice arrangement and increased grain boundaries post-OER (Fig. 10h). The presence of a  $d$  spacing of 0.217 nm, indexed to the (200) facet of  $CoOOH$ , further confirms the facilitated structural transformation by the introduced  $V_O$  during the OER. Subsequent correlative *operando* characterization and electrokinetic analyses suggest that a moderate  $V_O$  density can alter the O–O bond

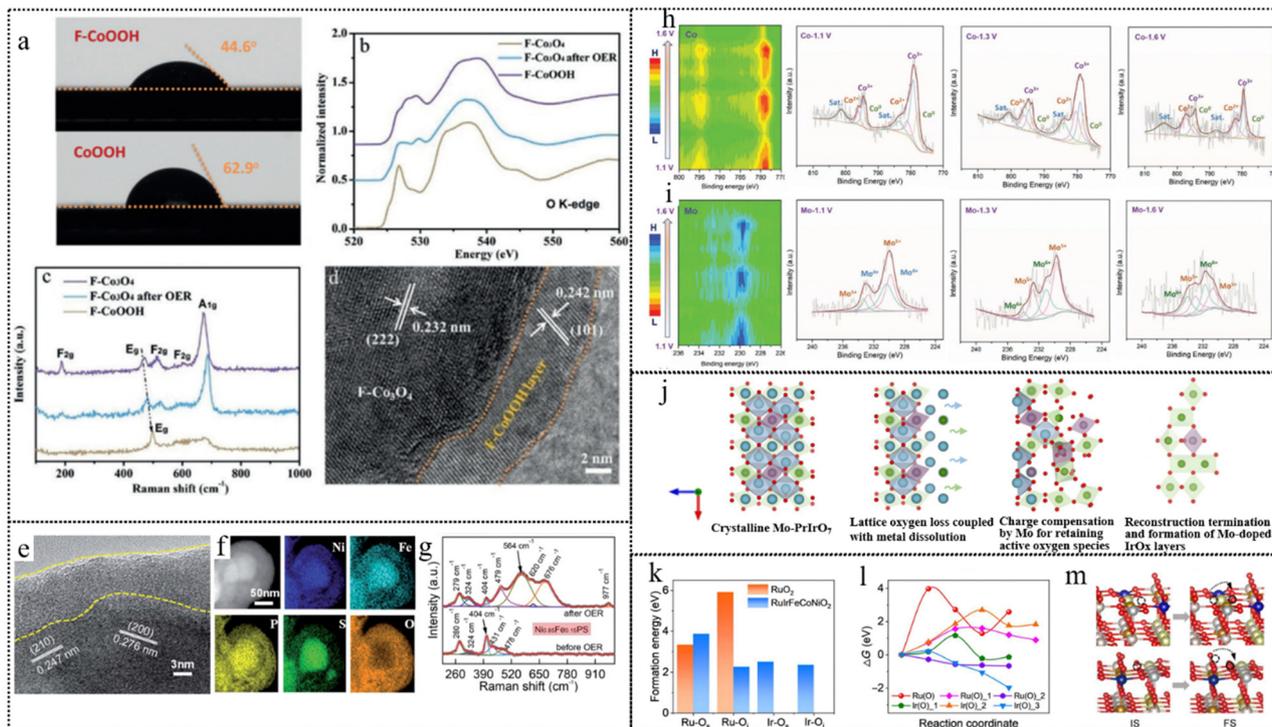
formation pathway, favoring an intramolecular nucleophilic attack pathway over a water nucleophilic attack pathway, which is more kinetically favorable for water oxidation.

Likewise, the presence of cation-type vacancies in precatalysts can stimulate rapid surface reconstruction and facilitate achieving reactive and stable photo/electrocatalysts. Peng *et al.* presented a method to improve the activity and stability of NiFe-LDH materials for the oxygen evolution reaction (OER) in alkaline media by atomic cation-vacancy engineering, which inhibits the dissolution of metals and allows the pre-catalyst to rapidly restructure into a structure with high catalytic activity.<sup>119</sup> The authors synthesized NiFe-LDH with  $Zn^{2+}/Al^{3+}$  doped precursors by a hydrothermal method and then selectively etched  $Zn^{2+}$  and  $Al^{3+}$  with 1.0 M KOH solution to construct  $Zn^{2+}$  vacancies,  $Al^{3+}$  vacancies, or mixed with  $Zn^{2+}$  and  $Al^{3+}$  vacancies on the basal plane of LDH nanosheets. The authors investigated the structural evolution of the LDH catalysts during the OER by CV (Fig. 10i) and Raman spectroscopy (Fig. 10j and k). They show that the cation vacancies (especially cation(III) vacancies) expedite the formation of  $\gamma$ -(NiFe)OOH active sites from NiFe(OH)<sub>2</sub> layers in the LDHs, and also change the structure evolution pathway from a direct conversion to a two-step conversion *via* the  $\beta$ -(NiFe)OOH intermediate (Fig. 10l). The researchers also show that the cation vacancies lower the free energy for deprotonation on the LDH surfaces, thus facilitating the OER process.

Another similar example is that He *et al.* reported Ni(OH)<sub>2</sub> with controllable concentrations of nickel vacancies ( $V_{Ni}$ ) that promote the formation of active species (NiOOH) by enhancing charge transfer and reducing formation energy.<sup>120</sup> The photo/electrochemical measurements and DFT calculations support the correlation between  $V_{Ni}$  and the catalysts' reconstruction ability during reactions. Specifically, the introduction of  $V_{Ni}$  leads to the emergence of new defect levels and an increase in hole density near the Fermi level, accelerating charge transfer and facilitating the generation of NiOOH.

Ion incorporation into pre-catalysts, as another type of conventional point defect, optimizes their electronic structure and promotes dynamic restructuring.<sup>153</sup> This reconstruction is induced by doping with metal or non-metal heteroatoms, triggering a reorganization of atomic arrangements.<sup>154,155</sup> Specifically, doping of F on the metal-based catalysts can form weak M–F bonds. These bonds are readily broken during the reconstruction phase, leading to the conversion into F-doped metal (oxy)hydroxides. Furthermore, doping of F adjusts the electronic configuration of metal sites and modulates the energy of adsorption, thereby enhancing catalytic activity.<sup>156</sup> Chen *et al.* exploited the high electronegativity and weak metal–fluorine bond characteristics of fluorine anions, which enabled them to migrate dynamically from inside to outside and enrich on the surface of cobalt-based materials during electrochemical activation.<sup>121</sup> The surface-enriched fluorine anions increased the hydrophilicity of the materials, facilitating the contact between reactants and active sites, as well as improving the photon/electron transfer ability and adsorption energy of reactants. The researchers conducted a series of structural





**Fig. 11** (a) Static water contact-angle measurements of F-CoOOH and CoOOH products. (b) The O K-edge XANES spectra and (c) Raman spectra of F-CoOOH, F-Co<sub>3</sub>O<sub>4</sub>, and F-Co<sub>3</sub>O<sub>4</sub> after the OER test. (d) The *ex situ* HRTEM image of the F-Co<sub>3</sub>O<sub>4</sub> product after OER tests. This figure has been reproduced from ref. 121 with permission from Wiley-VCH, copyright 2018. (e) HRTEM and (f) TEM-EDX maps of the Ni<sub>0.85</sub>Fe<sub>0.15</sub>PS after catalysis. (g) Raman spectra of Ni<sub>0.85</sub>Fe<sub>0.15</sub>PS before and after the OER. This figure has been reproduced from ref. 122 with permission from The Royal Society of Chemistry, copyright 2020. (h) and (i) Potential-dependent XPS structural evolution of (h) Co 2p and (i) Mo 3d in the CMN-500 catalyst at the reconstruction stage within the potential ranging from 1.1 to 1.6 V during the alkaline OER, and the corresponding fitted XPS spectra at the potentials of 1.1, 1.3, and 1.6 V. This figure has been reproduced from ref. 123 with permission from Wiley-VCH, copyright 2021. (j) Schematic illustration of chemical surface reconstruction over Mo-PrIrO<sub>7</sub>. This figure has been reproduced from ref. 157 with permission from Springer Nature Publishing Group, copyright 2023. (k) Formation energy of O vacancies in RuO<sub>2</sub> and RuIrFeCoNiO<sub>2</sub>. O<sub>s</sub> and O<sub>i</sub> refer to the surface and inner oxygen that coordinate with Ru/Ir, respectively. (l) Free energy diagram for oxygen diffusion of Ru-coordinated oxygen in RuO<sub>2</sub> [Ru(O)] and RuIrFeCoNiO<sub>2</sub> [Ru(O)<sub>1</sub>, and Ru(O)<sub>2</sub>], and Ir-coordinated oxygen in RuIrFeCoNiO<sub>2</sub> [Ir(O)<sub>1</sub>, Ir(O)<sub>2</sub>, and Ir(O)<sub>3</sub>]. (m) Representative models for IS and FS of Ru(O)<sub>2</sub> (top) and Ir(O)<sub>3</sub> (bottom) in I, the hazel, pale green, tan, cyaneous, gray, and red spheres in the atomic models represent Ru, Ir, Fe, Co, Ni, and O atoms, respectively. IS and FS represent the initial state and final state, respectively. This figure has been reproduced from ref. 124 with permission from American Association for the Advancement of Science, copyright 2023.

characterization studies to investigate the effect of F<sup>-</sup> on the structure reconstruction of F-Co<sub>3</sub>O<sub>4</sub> under OER conditions. As shown in Fig. 11a, F<sup>-</sup> are primarily concentrated on the surface of F-CoOOH nanosheets, and the surface hydrophilicity increased after fluorine doping, which facilitated the contact between water and active sites. The synthesis of the pure phase of Co<sub>3</sub>O<sub>4</sub> was revealed from the XRD pattern of the F-Co<sub>3</sub>O<sub>4</sub> sample post the OER test. The O K-edge XANES spectra of F-Co<sub>3</sub>O<sub>4</sub> after OER tests (Fig. 11b) indicated the evolution of surface oxyhydroxides during the OER process, further supported by the Raman spectra of F-Co<sub>3</sub>O<sub>4</sub> before and after OER tests (Fig. 11c). The missing F<sub>2g</sub> peak and the rightward shift of E<sub>g</sub> and A<sub>1g</sub> peaks suggested a gradual conversion of the surface of F-Co<sub>3</sub>O<sub>4</sub> into F-CoOOH during the OER process. In Fig. 11d, the *ex situ* HRTEM images and element mapping revealed the visual surface reconstruction phenomenon. These results suggested that F anions can dynamically migrate within the material to its surface during the photo/electrochemical activation process. This migration promotes the generation of

catalytically active layers (F-CoOOH) and enhances the OER catalytic activity of cobalt-based electrode materials.

Dopants can also enhance charge transfer and modulate the electronic structure in pre-catalysts, leading to changes in the oxidation state of metal atoms and influencing the kinetics of reconstruction. The team of Peng *et al.* found that a certain content of Fe doping in NiPS (Ni<sub>0.85</sub>Fe<sub>0.15</sub>PS) stimulates the activation of higher oxidation state NiFe (oxy)hydroxides, driving the irreversible surface reconstructing.<sup>122</sup> The surface morphology and composition of Ni<sub>1-x</sub>Fe<sub>x</sub>PS compounds before and after the OER by HRTEM (Fig. 11e) and TEM-EDX maps (Fig. 11f) revealed that moderate Fe regulation in NiPS forms distortion on MOOH@M<sub>x</sub>SO<sub>4</sub>@M<sub>x</sub>PO<sub>4</sub> surfaces during the OER process. The authors used Raman spectroscopy to study the vibrational modes of the Ni<sub>1-x</sub>Fe<sub>x</sub>PS compounds (Fig. 11g) and found that Fe doping enhances the electron transfer between Ni and S atoms, which is beneficial for self-reconstruction in the OER. DFT calculations were conducted to investigate the electronic structure and mechanism of the Ni<sub>1-x</sub>Fe<sub>x</sub>PS compounds



and found that new active sites formed by Fe doping after structure reconstruction exhibit higher nucleophilicity compared to Ni sites, rendering them more active for the oxygen evolution reaction (OER). This enhanced activity is attributed to the facile oxidation state change between Fe(II) and Fe(III).

Some dopant elements are relatively unstable under catalytic conditions, leading to reconstruction due to reactions, such as some main group elements (N, S, Se, P, *etc.*<sup>34,158,159</sup>) or transition metal elements containing multiple oxidation state forms. Bai *et al.* employed a cationic molybdenum-substitution strategy to prepare polymorphic cobalt with various phase compositions by a solid-state reaction on two-dimensional cobalt oxide (CoO) nanosheets. These compositions include CoO and a metallic cobalt phase with a face-centered cubic structure or hexagonal close-packing (hcp).<sup>123</sup> The Mo substitution acts as an electron donor, stabilizing the metastable hcp metallic cobalt phase in polymorphic cobalt oxides. This process reduces the energy barrier of rapid surface changes, a critical factor in the generation and stabilization of active sites for the OER. The authors used the real-time XPS (Fig. 11h and i) technique to track the evolution of surface chemical states during the catalytic reaction, revealing the effects of different phase compositions on the initial surface reconstruction kinetics under OER conditions in an alkaline environment. The results showed that the hcp metallic cobalt phase in the molybdenum-promoted polymorphic cobalt catalyst could effectively facilitate surface reconstructing and generate durable active sites, thus reaching optimal OER catalysis performance. According to DFT calculations, the most rapid structural reconstruction kinetics and the lowest free energy barrier among the investigated catalysts are attributed to the co-existence of the mixed metallic phases and oxide phases of cobalt, as well as the presence of molybdenum and nitrogen.

Controlling reconstruction in a precise manner is pivotal for fabricating highly active catalysts, especially for noble metal catalysts with high activity and unsatisfactory stability, to prevent the excessive dissolution of active sites. Chen *et al.* presented a novel catalyst, orthorhombic Pr<sub>3</sub>IrO<sub>7</sub> doped with high-valence Mo, which can catalyze water oxidation reactions under acidic conditions.<sup>157</sup> The authors use density functional theory calculations and experimental methods to reveal the process of Mo-promoted surface reconstruction depicted in Fig. 11j, forming self-terminated Ir-Obri-Mo active species. Mo doping not only enhances the Ir–O covalency and lattice oxygen activity, leading to faster surface reconstruction but also buffers the charge imbalance caused by Pr leaching by increasing the Mo oxidation state, preventing excessive loss of lattice oxygen. TEM and XPS after photo/electrochemical cycling show that Mo doping can trigger deeper surface reconstruction with more exposure of catalytically active sites.

Uncontrollable reconstruction results in the instability of catalysts particularly severe in acidic water electrolysis. Grain boundary engineering (GBE) of a microstructure has been demonstrated to enhance the intergranular corrosion,<sup>160</sup> oxidation,<sup>161</sup> and other mechanical properties in various industrially relevant alloys, which shed light on increasing

the stability of catalysts and rationally controlling the reconstruction. Hu *et al.* presented a method to design efficient OER catalysts by combining various components and grain boundaries (GB) which overcome the thermodynamic solubility limits of different metal elements, enhancing the catalyst's performance.<sup>124</sup> The catalyst was prepared by a fast, nonequilibrium salt melt synthesis method, which introduced acid-resistant elements (Ir) and catalytically active elements (Fe, Co, and Ni) into the RuO<sub>2</sub> lattice, forming a nanosheet structure with abundant grain boundaries (GBs). The authors show that the catalyst undergoes a structural evolution from RuO<sub>2</sub> to hydrous RuO<sub>x</sub> and IrO<sub>x</sub> species under anodic potentials by using *operando* Raman spectroscopy. This structural transformation is responsible for the high activity and durability of the catalyst. The oxygen-forming energy (Fig. 11k) and diffusing energy barrier (Fig. 11l) of the O atom coordinated by Ru/Ir were calculated by DFT. The oxygen diffusing energy barrier of RuIrFeCoNiO<sub>2</sub> is much lowered so that Ir/Ru tend to attract the O coordination from the surrounding Fe/Co/Ni atoms. The proposed scenario (Fig. 11m) depicted by DFT calculations shows that the defects O vacancies generated near the Ru/Ir atoms can be readily repaired by the diffusion of the neighboring Fe/Co/Ni coordination O atoms, prolonging catalyst durability. It can be summarized as that the incorporation of foreign elements and GB affects the charge redistribution, lowers the formation energy and diffusion energy barrier of O vacancies, and enhances the stability of surface oxygen.

### 3.3 Heterostructure constructions

Heterostructures that break both space-inversion and time-inversion symmetries provide opportunities to manipulate electronic and structural properties.<sup>162–164</sup> Additionally, interfacial reconstruction, which depends closely on the symmetry breaking at the interface, is considered one of the primary driving forces behind the reconstruction of more functional structures.<sup>165–168</sup>

Heterostructures, through their intricate interfaces, play a pivotal role in fostering the generation of active species. These interfaces serve as accelerators for photon/electron transfer, particularly in managing defects within these interface regions. Moreover, the robust interfacial electronic interactions exert a significant influence by reshaping the electronic configuration of reactive species. This modification effectively unveils additional active sites within pre-catalysts, thereby facilitating their reconstruction for enhanced reactivity. Furthermore, the localized bonding environments established within heterostructures play a crucial role in expediting the conversion of species, ensuring a swift and efficient catalytic process. For example, Tao *et al.* reported the synthesis and characterization of a novel heterostructure catalyst, NiSe<sub>2</sub>/FeSe<sub>2</sub>-NC, for the OER, achieving mutual advantages after reconstruction.<sup>125</sup> By using *in situ* Raman spectroscopy (Fig. 12a) and *operando* XAFS, the authors show that the catalyst undergoes dynamic surface reconstruction during the OER, forming a Ni–O–Fe bonding configuration with both Ni and Fe in a trivalent state derived from the fitting results of XAFS (Fig. 12b), which act as the real



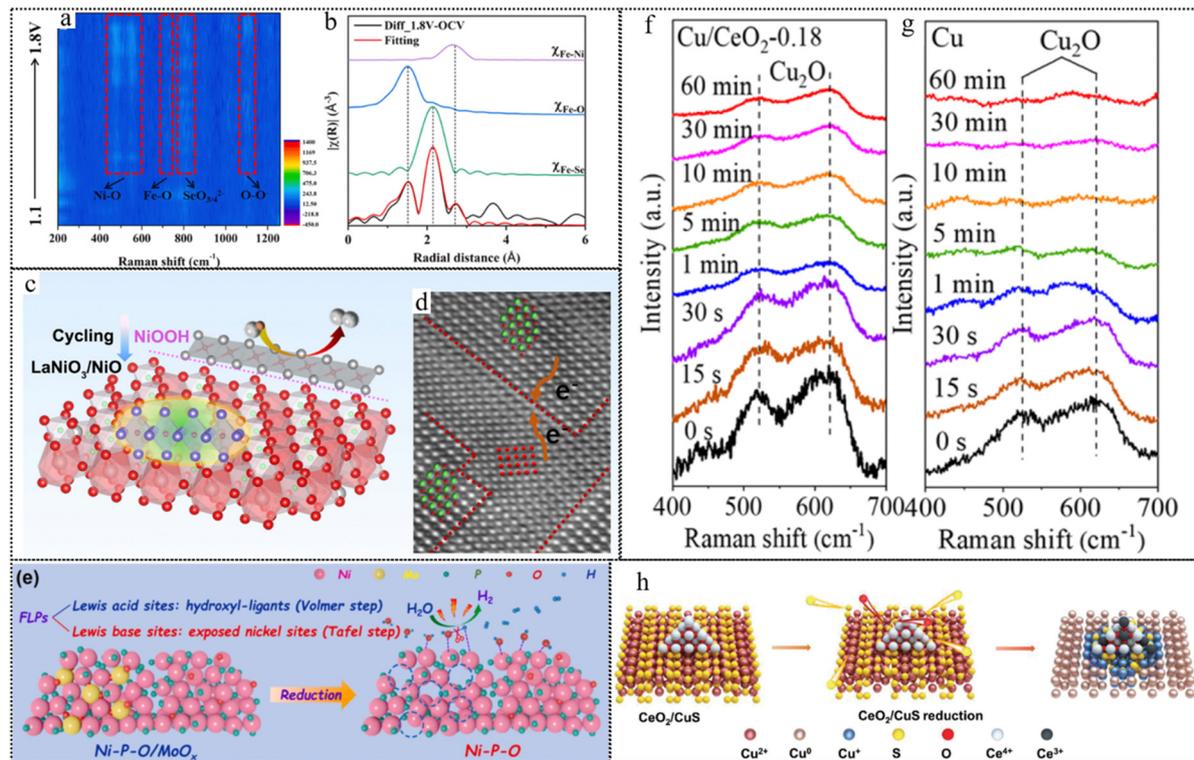


Fig. 12 (a) Raman intensity maps of NiSe<sub>2</sub>/FeSe<sub>2</sub>-NC. (b) Fitting results of diff\_1.8V-OCV and contributions from different paths. This figure has been reproduced from ref. 125 with permission from Elsevier, copyright 2023. (c) Schematic illustration of electrochemical heterostructure reconstruction. (d) Atomic HAADF-STEM image of LaNiO<sub>3</sub>/NiO. This figure has been reproduced from ref. 126 with permission from American Chemical Society, copyright 2022. (e) The mechanism of the electrochemical *in situ* reconstruction for Ni-P-O/MoO<sub>x</sub> during alkaline HER. This figure has been reproduced from ref. 128 with permission from Elsevier, copyright 2022. *Operando* Raman spectra of Cu/CeO<sub>2</sub>-0.18 (f) and Cu (g) evolution as a function of reaction time at open circuit potential (OCP). This figure has been reproduced from ref. 169 with permission from American Chemical Society, copyright 2022. (h) Schematic representation of the reconstruction process of CeO<sub>2</sub>/CuS. This figure has been reproduced from ref. 170 with permission from Wiley-VCH, copyright 2023.

active sites. They also show that the FeSe<sub>2</sub> nanodomains are responsible for OH<sup>-</sup> adsorption and forming O-O<sup>-</sup> intermediates, while the NiSe<sub>2</sub> compositions activate O-O coupling and facilitate the evolution of O<sub>2</sub>. DFT calculations illustrate that the heterojunction between NiSe<sub>2</sub>/FeSe<sub>2</sub> improves the electron conductivity and modulates the Se vacancy formation energy, which activates transformation of NiSe<sub>2</sub>/FeSe<sub>2</sub> into (oxy)hydroxides and maintains the stability of the main structure.

Another example is that Wei *et al.* reported a heterostructure constructed by deliberately making La deficiency in LaNO<sub>3</sub> perovskite. This heterogeneous structure can facilitate the formation of the active site and composition reconstitution, forming a distinct boundary between the perovskite host and separation of the NiO phase for the OER.<sup>126</sup> The authors found that the charge transfer on the interface causes the electron rearrangement of Ni ions, which leads to the *in situ* formation of an active NiOOH layer at a relatively low potential. The schematic process is illustrated in Fig. 12c and the heterostructure was confirmed by the atomic HAADF-STEM image (Fig. 12d). The LaNiO<sub>3</sub>/NiO heterostructure demonstrates a remarkable enhancement in OER activity and stability preventing La and Ni from dissolution under the OER conditions. The active centers and the surface reconstruction process of the

LaNiO<sub>3</sub>/NiO heterostructure were investigated by *operando* Raman spectroscopy, quasi *in situ* XPS, and DFT calculations. They showed that the Ni sites at the interface are the main active sites for the OER, which can completely transform into NiOOH at a potential of ~1.35 V vs. RHE. They also illustrated the photon/electron transfer path and delineated various oxidation states across different regions of the LaNiO<sub>3</sub>/NiO heterostructure through cyclic voltammetry (CV) analysis.

In certain heterostructure systems, heterostructural components that are initially inactive may transform functional species, thereby enhancing the catalytic activity by providing support to the active species. Liao *et al.* reported a novel method to create a multisite catalyst for enhancing the hydrogen evolution reaction (HER) on a nano-Pd surface in alkaline media.<sup>127</sup> The catalyst is composed of core/shell Pd/FeO<sub>x</sub>(OH)<sub>2-2x</sub> nanoparticles (NPs), where the FeO<sub>x</sub>(OH)<sub>2-2x</sub> shell acts as a water dissociation site and the Pd core acts as a hydrogen formation site. The coverage of the FeO<sub>x</sub>(OH)<sub>2-2x</sub> shell on the Pd surface can be tuned by chemical cycling, which converts the inert Fe<sub>3</sub>O<sub>4</sub> shell to FeO<sub>x</sub>(OH)<sub>2-2x</sub> and controls its removal. There remain challenges and limitations of existing non-noble alkaline HER catalysts impeded by poor intrinsic activity and inert catalytic sites. To deal with this issue, Xiao



*et al.* synthesized a Ni–P–O/MoO<sub>x</sub> catalyst which consists of Ni(PO<sub>3</sub>)<sub>2</sub>–Ni<sub>2</sub>P nanoparticles coupled with MoO<sub>x</sub> on NiMoO<sub>4</sub> nanorods for the alkaline hydrogen evolution reaction.<sup>128</sup> The proposed mechanism of the chemical *in situ* reconstruction is illustrated in Fig. 12e. The catalyst undergoes structural evolution during the HER, as MoO<sub>x</sub> dissolves and exposes more nickel sites, and hydroxyl-ligands form on the surface of the nanoparticles according to the *ex situ* FTIR spectra and photo/electrochemical measurements. The DFT calculations reveal that the Ni–P–O catalyst integrates artificial frustrated Lewis pairs (FLPs) consisting of nickel sites as Lewis base sites and hydroxyl-ligands as Lewis acid sites, which can synergistically facilitate the Volmer and Tafel steps of the HER. The hydroxyl-ligands can strongly interact with OH\* and lower the energy barrier for water dissociation, while the nickel sites can optimize the H\* binding and desorption ability.

Notably, if reconstruction is not properly controlled, it may lead to an outrageous disintegration of active sites and degradation of the catalyst. However, the destructive effect of reconstruction can be mitigated by incorporating robust components. NiFe LDHs have garnered significant interests as noble-metal-free photo/electrocatalysts for the oxygen evolution reaction (OER) due to their abundant reserves and superior performance.<sup>171</sup> However, their stability requires further enhancement. In a recent study, Yang *et al.* addressed this challenge by constructing a heterostructure that combines the following components: outstanding activity but unsatisfactory durability NiFe LDH and good structure stability but insufficient activity birnessite (Bir).<sup>172</sup> The increased interlayer distance observed in XRD, along with the consequent relaxation of the proton bonding as indicated by the <sup>1</sup>H solid-state NMR results of LDH Bir, would promote rapid electrolyte substance transport. This, in turn, would mitigate the localized pH decrease that typically triggers the loss of active sites. As a result, the catalyst remains exceptionally stable even when working under high current densities. In another case in CO<sub>2</sub> reduction (CO<sub>2</sub>R), the specific structure of the catalyst is crucial for the selectivity of high-value-added products (C<sub>2+</sub>). Some literature indicates that the inherent characteristics of mono-metallic Cu restrict the selectivity for C<sub>2+</sub>.<sup>173–175</sup> Therefore, reconstruction from Cu with a high selective structure during the catalytic process is often undesirable.

Wang *et al.* reported a novel synthesis of a copper/ceria heterostructure that can enhance the electrocatalytic conversion of carbon dioxide (CO<sub>2</sub>) to multicarbon (C<sub>2+</sub>) products.<sup>128</sup> The authors induce strain in the Cu domain through the introduction of a ceria shell while simultaneously preventing structural changes in copper, which keeps favorable selectivity of the CO<sub>2</sub>R. The authors characterize the structure, morphology, and chemical state of the catalysts by XRD, SEM, TEM, EDS, XPS, AES, and XAS. The authors confirm that the dominant oxidation states for Cu and Ce are Cu<sup>+</sup> and Ce<sup>4+</sup>, respectively. After analyzing the CO<sub>2</sub>R by HRTEM, XRD, and ATR-SEIRAS, the results reveal the intergrowth of Cu and CeO<sub>2</sub> and that the cubic structure was preserved. From the *in situ* Raman spectra in Fig. 12f and g, after applying a cathodic potential

(–0.7 V vs. RHE), the characteristic peaks of Cu<sub>2</sub>O gradually declined within 10 minutes for the pure Cu<sub>2</sub>O, indicating its complete reduction to Cu. However, in the Cu<sub>2</sub>O/CeO<sub>2–0.18</sub> heterostructure, the characteristic Cu<sub>2</sub>O modes were retained for over 60 minutes, indicating enhanced durability of Cu<sup>+</sup> under cathodic potential. Analysis of the Raman results suggests that Cu<sub>2</sub>O within the heterostructure exhibits greater resistance to cathodic reduction, attributed to the interaction between Cu<sub>2</sub>O and CeO<sub>2</sub>, favoring the formation of strained copper domains. A similar phenomenon exists in CuS/CeO<sub>2</sub> reported by Yan *et al.*<sup>170</sup> By using various *in situ* and *ex situ* techniques, such as XAS, Raman spectroscopy, FTIR, and DFT calculations. The authors propose that the presence of CeO<sub>2</sub> inhibits the complete reduction of CuS and preserves the active Cu<sup>0</sup>/Cu<sup>+</sup> species illustrated in Fig. 12h, which reduces the energy barrier of C–C coupling and increases the selectivity of C<sub>2+</sub>. From the analysis of XPS, the increase in the Ce<sup>3+</sup> concentration in CeO<sub>2</sub>/CuS after CO<sub>2</sub>RR explains the stability of Cu<sup>+</sup> in CeO<sub>2</sub>/CuS, which means CeO<sub>2</sub> is used as a self-sacrificing agent. The heterostructure of CuS/CeO<sub>2</sub> successfully undermines the undesired reconstruction from Cu<sup>+</sup> to metallic Cu during the CO<sub>2</sub>RR.

### 3.4 Leaching

Reconstructions resulting from leaching processes encompass various distinct categories within material structures. Firstly, these include instances where unstable ions, such as specific sites within the perovskite structure contribute to leaching phenomena. Additionally, spontaneous leaching can be triggered by alkali metal ions or halide ions within the material matrix. Under catalytic conditions, pre-catalysts may undergo chemical reactions leading to leaching, commonly observed under oxidative conditions. Moreover, the interaction between the catalyst surface and the electrolyte engenders dynamic leaching and deposition processes, contributing to the evolving catalytic dynamics. These categories encapsulate the diverse mechanisms through which leaching induces reconstructions across different material systems, particularly in catalytic scenarios.

In the majority of inorganic ABO<sub>3</sub> or AB(OH)<sub>6</sub> perovskites, alkaline metals and lanthanides typically fill the A-site, while a variety of transition metals such as Ni, Co, and Ir are found at the B-site.<sup>176</sup> The process of reconstruction can lead to the leaching of A or B-site cations, which in turn can trigger the formation of active species.<sup>177</sup> These include active OH<sup>–</sup> groups and reactive oxygen ligands, which can enhance their OER performance. Tang *et al.* presented a novel perovskite catalyst that can accelerate the surface reconstruction process by combining the leaching atoms and the oxyanions on the solid-liquid interface, thereby enhancing the oxygen evolution reaction efficiency.<sup>129</sup> The spontaneous leaching of La into electrolytes, leaving plenty of vacancies, triggered the surface reconstruction as shown in Fig. 13a. The authors found that adding oxyanions such as sulfate or phosphate to the perovskite catalyst can reduce the adsorption energy of surface hydroxyls, facilitate the formation and migration of oxygen

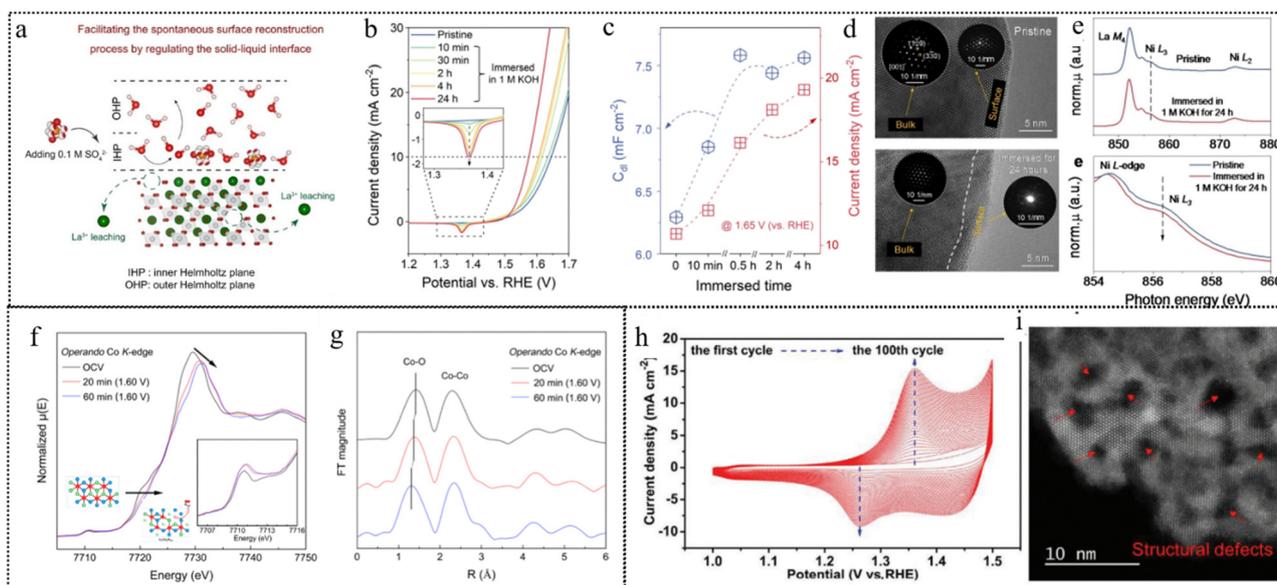


vacancies, and enhance the lattice oxygen oxidation mechanism. From Fig. 13b and c, the increase in the reduction peak area over time of immersion, coupled with a simultaneous enhancement in both OER performance and electrochemically active surface area (ECSA), suggest the generation of additional active sites during the immersing process of  $\text{LaNiO}_{3-\delta}$ . The HRTEM images revealed an amorphous layer generated on the surface while the bulk structure remained unchanged when soaked in a 1 M KOH solution for 24 hours. Based on the XAFS and XPS characterization (Fig. 13e), the reconstructed layer presented that the oxidation state of Ni reduced from 3+ to 2+. The results showed that the perovskite catalyst treated with sulfate had the best oxygen evolution reaction activity and stability.

Alkali metal elements such as Li/Na/K and some halide elements such as F/Cl possess quite large reduction/oxidation potential, which are stable in their ion forms. When the layered materials contain the above elements, they tend to spontaneously dissolve to promote material reconstruction. Li *et al.* found that spinel  $\text{Li}_2\text{Co}_2\text{O}_4$  (LCO) undergoes a spontaneous delithiation process that triggers the formation of an amorphous layer of  $\text{Li}_x\text{Co}_2\text{O}_{4-\delta}(\text{OH})_\delta$  on the surface.<sup>130</sup> In Fig. 13f, the *operando* XAS data show that the valence state of Co ions increases from 3+ to 3.3+ and the  $\text{CoO}_6$  octahedron became distorted under OER conditions. The EXAFS fitting data (Fig. 13g) also show that the average Co–O bond distance decreased and the Co–Co distance become disordered. These results suggest that the surface structure of LCO undergoes a dynamic reconstruction under OER conditions. The *ex situ* XRD

and Raman spectra show that the overall crystal phase of LCO remains unaltered after CV scans, while the TEM and HAADF-STEM images reveal an amorphous structure on the surface area. The post-mortem XAS data demonstrate an increase in the concentration of  $\text{Co}^{4+}$  ions alongside the CV scans, as evidenced by the positive energy shift and the emergence of a peak at the Co  $L_{2,3}$ -edge. The O K-edge XAS data show that the oxygen valence state also increases, as indicated by the increased intensity and the decreased energy position of the pre-edge peaks. The authors identify the simultaneous occurrence of  $\text{Co}^{4+}$  and oxidized O ions with the amorphous active layer accompanied by a spontaneous delithiation process that reduces the Li/Co ratio from 2 to 0.8, as confirmed by ICP-OES analysis. Another example is that Cui *et al.* used electrochemical activation to intentionally extract Li ions from  $\text{LiCoO}_2$ .<sup>178</sup> The delithiation process induces a phase transition from hexagonal to monoclinic  $\text{Li}_{0.5}\text{CoO}_2$ , which alters the oxidation state and covalency of Co and O atoms and makes the material more electrophilic and metallic.

Song *et al.* synthesized cobalt oxychloride ( $\text{Co}_2(\text{OH})_3\text{Cl}$ ) as the OER precatalyst.<sup>131</sup> The removal of the  $\text{Cl}^-$  lattice during the chemical process facilitates continuous reconstruction and enhanced OER activities. The  $\text{Co}_2(\text{OH})_3\text{Cl}$  precatalyst shows an ever-increasing chemical surface area during the CV activation (Fig. 13h), indicating more active sites exposed for the reaction. The etching of the  $\text{Cl}^-$  lattice is detected by the  $\text{AgNO}_3$  aqueous solution and the XPS spectra, suggesting the irreversible cleavage of the Co–Cl bond in  $\text{Co}_2(\text{OH})_3\text{Cl}$ . The STEM images of activation-derived (AC)- $\text{Co}_2(\text{OH})_3\text{Cl}$  in Fig. 13i showed lots of



**Fig. 13** (a) The proposed mechanism for surface reconstruction immersed in electrolytes with oxyanions. (b) Backward scan polarization curves (1.2–1.7 V vs. RHE) of  $\text{LaNiO}_{3-\delta}$ . (c) The variation of current density (at 1.65 V vs. RHE) and double layer capacitance ( $C_{dl}$ ) with immersed time. (d) STEM of pristine (upper)  $\text{LaNiO}_{3-\delta}$  being immersed in 1 M KOH for 24 hours (lower). (e) XAFS spectra of Ni  $L_{2,3}$ -edges, this figure has been reproduced from ref. 129 with permission from Wiley-VCH, copyright 2023. (f) Co K-edge XANES of  $\text{Li}_2\text{Co}_2\text{O}_4$  electrodes with open circuit voltage (OCV) and at 1.60 V vs. RHE after 20 and 60 min in 1 M KOH. The inset shows the pre-edge peaks. (g) FT of the corresponding Co K-edge EXAFS. This figure has been reproduced from ref. 130 with permission from American Chemical Society, copyright 2019. (h) CV curves of 100 cycles in 1.0 M KOH with a scan rate of  $100 \text{ mV s}^{-1}$ . (i) STEM images of (AC)- $\text{Co}_2(\text{OH})_3\text{Cl}$ . This figure has been reproduced from ref. 131 with permission from Wiley-VCH, copyright 2019.



defects. The *operando* XANES and EXAFS spectra reveal that the reconstruction-derived component has a similar structure framework to the  $\beta$ -CoOOH bulk, but with lower intensity of the white peak and higher intensity of the pre-edge peak, indicating structure distortion and the existence of defects consistent with the STEM results. The EXAFS spectra also show that the surface reconstruction-derived component has lower coordination numbers of the Co–O and Co–Co bonds than the  $\beta$ -CoOOH core part, demonstrating unsaturated sites for potential catalysis.

Many materials undergo reconstruction under catalytic conditions, especially at high oxidation potentials, due to the dissolution of some of their constituent elements. For instance, metal alloys can be partially dissolved by forming high-valent oxides, hydroxides, or oxyhydroxides. Similarly, transition metal compounds with main group elements (borides, carbides, nitrides, phosphides, and chalcogenides) can lose some of their anionic components by oxidation, leading to different degrees of dissolution. Moreover, materials with C, N, and O elements as the backbone, such as MOFs and PBAs, can also experience oxidation and dissolution of their framework structures, which facilitates the reconstruction process.

For alloy, Yang *et al.* reported the synthesis of IrFeCoNiCu high-entropy alloy (HEA) nanoparticles by a microwave-assisted shock synthesis method as catalysts for the acidic oxygen evolution reaction (OER).<sup>132</sup> The HEA nanoparticles exhibit excellent activity and stability in 0.1 M HClO<sub>4</sub> electrolyte, with a lower overpotential and a smaller Tafel slope than pure Ir and other Ir-based bimetallic alloys. The authors analyzed the structural evolution of the HEA nanoparticles under OER conditions by STEM-EDX, XRD, XPS, and other techniques, and found that an enriched Ir shell layer generates on the surface of the HEA nanoparticles after chemical activation, mainly due to the dissolution of 3d metals (Fe, Co, Ni, and Cu). The schematic reconstruction pathway is illustrated in Fig. 14a. The core of the nanoparticles preserves the homogeneous uniform phase of the HEA structure without notable phase separation or elemental segregation. The shell layer has a polycrystalline structure and a thin oxide layer on the surface confirmed by STEM images in Fig. 14b and c. The authors suggest that this reconstructed structure is relatively durable and does not undergo further dramatic structural degradation during stability tests. This article indicated that the near-surface structure of HEA nanoparticles exhibits a certain degree of structural dynamics under acidic working conditions.

Ning *et al.* presented a method of preparing an efficient alkaline fresh water and seawater electrolysis catalyst based on photo/electrochemical reconstructing.<sup>133</sup> The catalyst consists of Fe-doped Ni<sub>0.8</sub>Ni<sub>0.2</sub>Mo<sub>0.8</sub>N microrods, which are grown on the Ni foam by hydrothermal and ammonia reduction nitridation methods. The catalyst presents excellent HER performance at low Fe doping levels, but poor HER performance at high Fe doping levels. When the anodic potential is applied, the catalyst undergoes rapid chemical reconstruction confirmed by a dramatic change in the oxidation peak of CV (Fig. 14d), in which most of the Mo dissolves into the electrolyte, while Ni and Fe

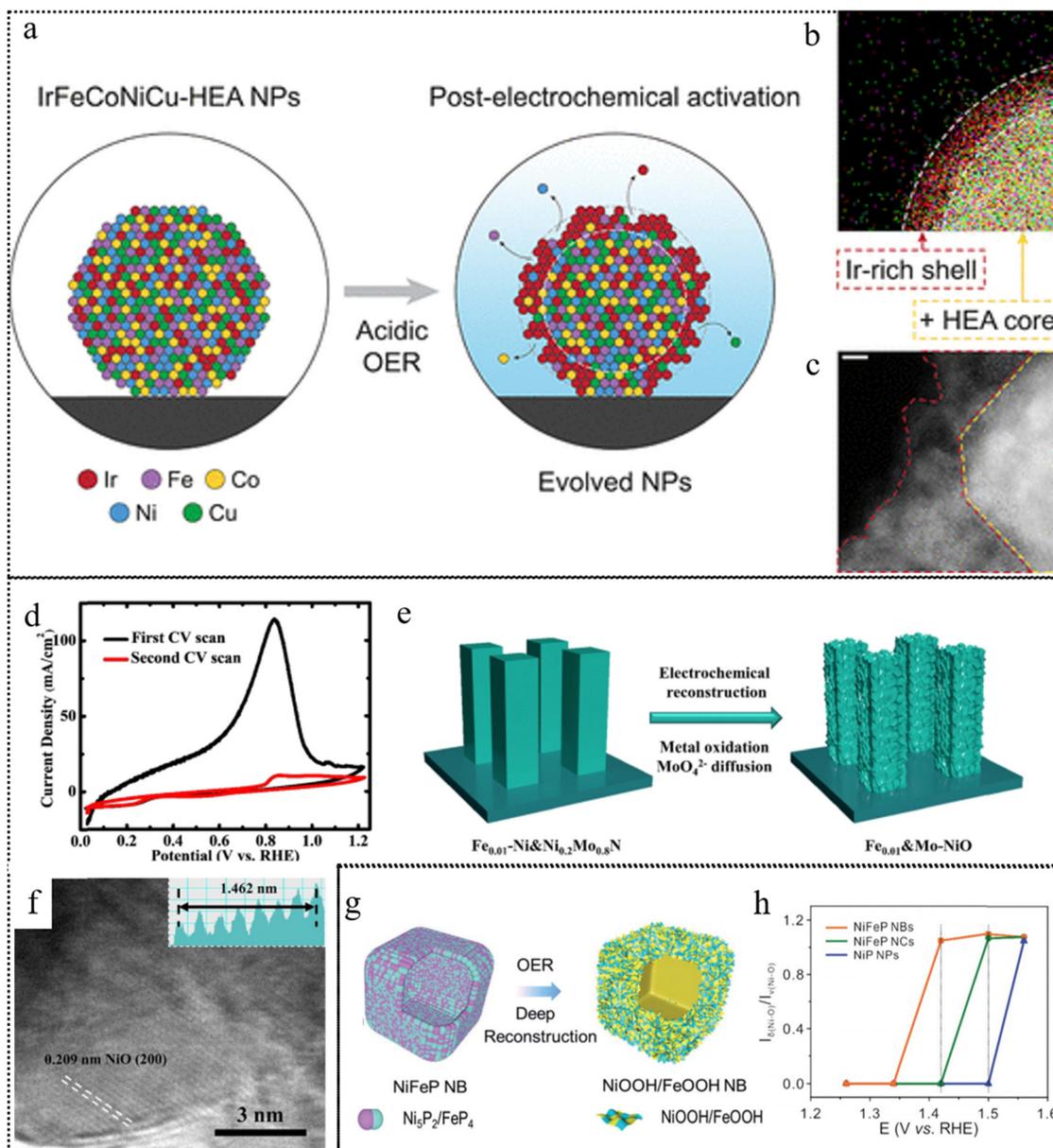
are oxidized to NiO and Fe<sub>2</sub>O<sub>3</sub> illustrated in Fig. 14f. The reconstructed catalyst consists of NiO co-doped with Fe and Mo (Fe<sub>x</sub>&Mo–NiO) as depicted in Fig. 14e and has comparable OER performance to the best OER catalysts. Therefore, the catalyst can serve as a bifunctional catalyst with both HER and OER activity.

By undergoing complete reconstruction from MOFs to (oxy)-hydroxides in alkaline media, MOFs have emerged as potential alternative precatalysts for the OER. Du and co-workers presented the synthesized Fe-doped Co-MOF nanosheets (Co-MOF/Fe) undergoing an entire transformation from MOFs to CoFe-LDHs in the KOH electrolyte, which are the true catalytically active catalysts for the OER.<sup>134</sup> Prussian blue analogs (PBAs) are also promising pre-catalysts for the OER, but most of them only undergo surface reconstruction and cannot completely transform into active metal (oxy)hydroxides. Yu *et al.* developed a chemical–physical synergistic etching strategy to enable the deep reconstruction of NiFe PBA nanocubes for efficient OER and Zn–air batteries.<sup>135</sup> They demonstrated that the NiFe PBA nanoboxes have hollow structures and dual vacancies (Ni vacancies and CN vacancies), which facilitate the deep reconstruction and preserve a certain amount of iron species.

Another work conducted by this group demonstrated that the NiFeP nanoboxes (NBs) can deeply reconstruct into defective NiOOH/FeOOH nanosheet assembled NiOOH/FeOOH NBs during the OER process, which exhibit outstanding OER performance with a low overpotential of 228 mV at 10 mA cm<sup>−2</sup> and excellent stability.<sup>136</sup> A schematic illustration is shown in Fig. 14g for the formation of NiFeP NBs and deep reconstruction of NiFeP NBs into hierarchical NiOOH/FeOOH NBs. By analyzing the bending vibration  $\delta(\text{Ni–O})$  and tensile vibration  $\nu(\text{Ni–O})$  peak values of NiOOH obtained from the Raman spectrum (Fig. 14h), the high ratio of  $\delta(\text{Ni–O})$  to  $\nu(\text{Ni–O})$  indicated  $\gamma$ -NiOOH formation, which is believed to have higher OER activity. It revealed that the NiFeP NBs deeply reconstructed into defective NiOOH/FeOOH nanosheet assembled NBs, while the NiFeP NCs only partially reconstructed into NiOOH/FeOOH NCs with Ni(OH)<sub>2</sub> remaining.

In addition, some elements leach out during the oxidation reaction and reassemble on the surface of the catalyst to form a polyanion layer. Constructing a polyanion layer as an electrostatic repulsion barrier within the active material surface not only effectively prevents Cl<sup>−</sup> adsorption but also does not compromise the exposure of active sites, which has been proven to effectively inhibit the CER reaction and achieve excellent stability of seawater electrolysis.<sup>33,179–181</sup> Dai *et al.* presented a novel anode material for water splitting in alkaline seawater.<sup>33</sup> The material consists of a NiFe hydroxide catalyst layer uniformly coated on a nickel sulfide (NiS<sub>x</sub>) layer, which is formed on porous nickel foam (NF) (abbreviated as Ni<sup>3</sup>). The material exhibits excellent catalytic activity and corrosion resistance in solar-driven alkaline seawater electrolysis, operating at industrially required current densities (0.4 to 1 A cm<sup>−2</sup>) for over 1000 hours. The authors found that during the anode activation process, the NiS<sub>x</sub> layer was oxidized to generate sulfate ions confirmed by the Raman spectra (Fig. 15a) and TOF secondary





**Fig. 14** (a) Illustration of the structural evolution of IrFeCoNiCu-HEA nanoparticle under acidic OER conditions. (b) STEM-EDX maps showing the Ir-rich shell layer near the surface of the nanoparticle, as highlighted by the dashed lines (scale bar: 10 nm). (c) Near-surface structure of the nanoparticle showing the HEA core, polycrystalline alloy shell layer ( $\sim 2\text{--}6$  nm), and surface oxide (indicated by white arrows) after electrochemical activation. This figure has been reproduced from ref. 132 with permission from American Chemical Society, copyright 2023. (d) CV scans and (e) Illustration of the electrochemical reconstruction for  $\text{Fe}_{0.01}\text{-Ni}\&\text{Ni}_{0.2}\text{Mo}_{0.8}\text{N}$ . (f) HRTEM image of  $\text{Fe}_{0.01}\&\text{Mo-NiO}$  after reconstruction. This figure has been reproduced from ref. 133 with permission from The Royal Society of Chemistry, copyright 2022. (g) Schematic illustration for the deep reconstruction of NiFeP nanoboxes into hierarchical NiOOH/FeOOH nanoboxes. (h) The relationship between the ratio of  $\delta(\text{Ni-O})$  to  $\nu(\text{Ni-O})$  and the applied potentials in the *in situ* Raman spectra. This figure has been reproduced from ref. 136 with permission from The Royal Society of Chemistry, copyright 2021.

ion mass spectrometry (TOF-SIMS) mapping (Fig. 15b), which intercalated into the NiFe layer along with carbonate ions from the alkaline solution, forming a polyanion-rich passivation layer that repelled and blocked chloride ions from seawater, thus enhancing the anode corrosion resistance.

In addition to element leaching, elements in the electrolyte also restructure and redeposit on the photo/electrocatalyst surface to form a dynamic equilibrium, especially the influence of Fe ions in the electrolyte on Ni, Co-based catalysts cannot be

ignored.<sup>182</sup> Kuai *et al.* reported on water oxidation reaction catalysts, mainly investigating the effects of iron incorporation and distribution in nickel hydroxide on its catalytic activity.<sup>137</sup> The authors also point out that Fe ions undergo dynamic exchange between the catalyst and the electrolyte under OER conditions, leading to chemical and structural changes in the catalyst surface. The authors used XRF microscopy as shown in Fig. 15c and XAS techniques to track the incorporation and distribution dynamics of iron in two-dimensional nickel



hydroxide nanosheets, and found that iron incorporation mainly occurred at the edge sites of nanosheets, and was closely related to the water oxidation reaction process. The authors further used DFT calculations to indicate that iron incorporation can enhance the reducibility of nickel sites, thereby improving the catalytic activity of water oxidation reactions. This research provides a comprehensive approach to understanding and tuning the dynamic changes of metal hydroxide photo/electrocatalytic interfaces, which is of great significance for controlling catalytic processes.

Moreover, Cuenya and colleagues used *in situ* electrochemical (scanning) transmission electron microscopy (EC-(S)TEM) combined with energy-dispersive X-ray spectroscopy (EDX) mapping to observe how Fe impurities in the electrolyte affect the surface structure and composition of nickel oxide (NiO) catalysts during the OER in alkaline solution.<sup>138</sup> The authors found that in a borate buffer solution containing 1 mM Fe, the NiO surface first formed a surface-rich NiFe double-layered hydroxide (NiFe-LDH) during the potential cycling process illustrated in the *operando* STEM image (Fig. 15d), which shifted the onset potential of the OER to the low potential direction. As shown in Fig. 15e, with the extension of the reaction time, when the LDH layer could no longer accommodate more Fe, Fe oxide (FeO<sub>x</sub>) aggregates began to appear on the surface, which reduced the active surface area of the catalyst.

Under a constant anodic potential, the LDH layer also gradually degraded, leading to more FeO<sub>x</sub> formation and partial Fe returning to the electrolyte. This process not only further reduced the performance of the catalyst but also did not restore NiO to its original state.

Even at a low operating current (10 mA cm<sup>-2</sup>), the leaching of Fe from the electrode surface is inevitable, and this phenomenon becomes more severe at a high current density (200 mA cm<sup>-2</sup>). Investigations on the dynamic restructuring processes of Fe include *in situ* observations,<sup>137,138</sup> as well as studies on the characteristics of Fe restructuring in Ni and Co systems. However, diverse perspectives exist regarding the description of the Fe restructuring process. The analysis of *operando* Mössbauer spectroscopy illustrated that a partial photo/electron transfer from Fe(III) occurs accompanied by the oxidation of Ni(II), generating higher oxidation states of Fe. These high-valent Fe species may significantly contribute to increasing OER activity. Subsequently, researchers employed *in situ* Mössbauer spectra to reveal the process of Fe(III) transforming into Fe(IV). The possibility of Fe(VI) has even been reported, as evidenced by cyclic voltammetry (CV) tests under non-aqueous conditions, indicating successive oxidation states of four, five, and six for NiFe LDH at elevated potentials. Researchers, such as Yanbo Li and colleagues proposed a self-healing mechanism involving Fe(VI) species.<sup>121</sup> In subsequent work, they successfully

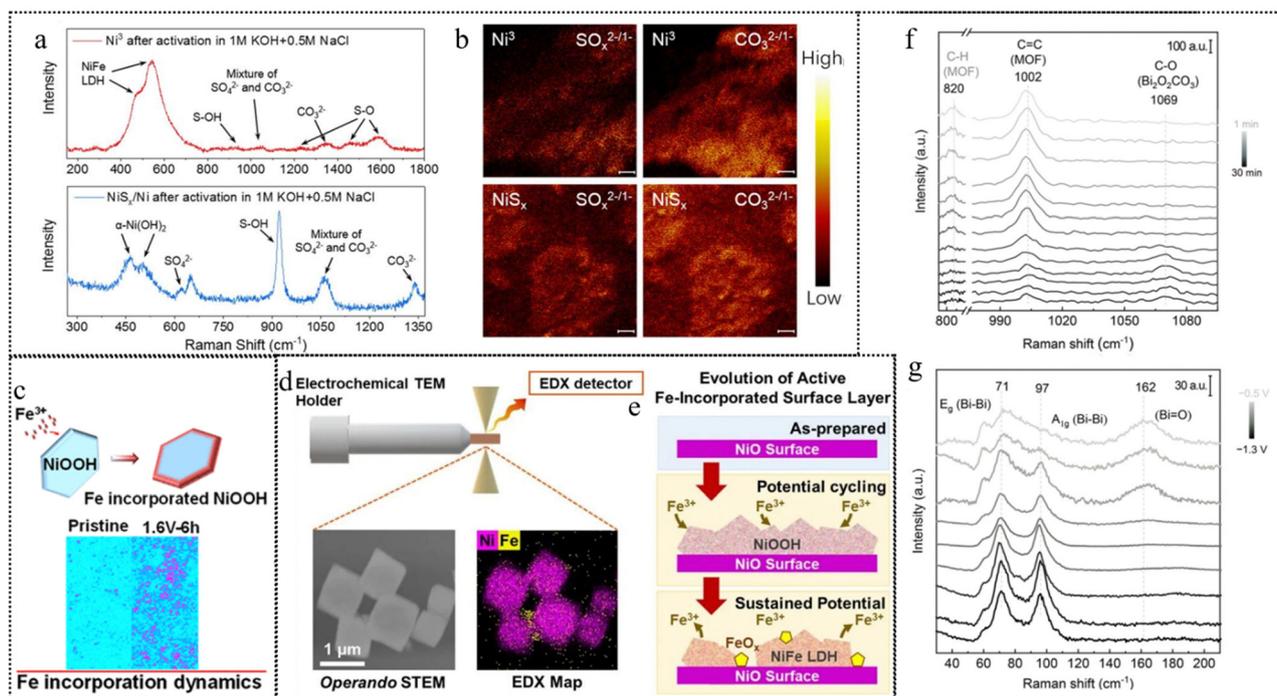


Fig. 15 (a) Raman spectra of Ni<sup>3+</sup> and NiS<sub>x</sub>/Ni after 12-h activation in 1 M KOH + 0.5 M NaCl. (b) TOF-SIMS map of SO<sub>x</sub><sup>2-/1-</sup> and CO<sub>3</sub><sup>2-/1-</sup> fragments from a Ni<sup>3+</sup> and NiS<sub>x</sub>/Ni electrode surface after activation in 1 M KOH + 0.5 M NaCl at 400 mA cm<sup>-2</sup>. This figure has been reproduced from ref. 33 with permission from National Academy of Science, copyright 2023. (c) Pristine Ni(OH)<sub>2</sub>, Ni(OH)<sub>2</sub> after 6 h CA measurement at 1.6 V vs. RHE in 1 M KOH containing 10 μM Fe<sup>3+</sup>. This figure has been reproduced from ref. 137 with permission from American Chemical Society, copyright 2021. (d) *In situ* STEM schematic setup and images and STEM/Fe EDX maps of NiO octahedra acquired in a 0.5 M borate buffer +1 mM Fe(NO<sub>3</sub>)<sub>3</sub> solution. (e) Illustration of the evolution of Fe incorporated surface reconstruction. This figure has been reproduced from ref. 138 with permission from American Chemical Society, copyright 2023. *In situ* Raman spectra revealing the conversion from (f) Bi-MOF NR to Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> NS on exposure to 0.1 M KHCO<sub>3</sub> and (g) Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> NS to Bi NS under CRR conditions. This figure has been reproduced from ref. 139 with permission from Wiley-VCH, copyright 2021.



enriched Fe(vi)O42-during continuous OER processes under aqueous conditions.<sup>121</sup>

Qiao *et al.* reported a comprehensive investigation into the controllable reconstruction of Bi-based MOFs (Bi-MOFs) for photo/electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR).<sup>139</sup> By analyzing the *in situ* Raman spectroscopy results in Fig. 15f and g, the band at 820 cm<sup>-1</sup> disappeared and a new band referred to the stretching vibration mode of C–O in Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> appeared after immersing in 0.1 M KHCO<sub>3</sub> for 20 min, which indicated this dynamic transformation from Bi-MOF to Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub>. When applying potential from –0.5 to –1.3 V vs. RHE, the typical vibration modes of C–O and Bi=O of Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> faded gradually, in the meantime two new bands of Bi–Bi appeared at 71 and 97 cm<sup>-1</sup>. The authors revealed that the reconstruction process can be separated into two steps: (1) electrolyte-regulated conversion of Bi-MOF nanorods (NR) to Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> nanosheets (NS) by bicarbonate-initiated ligand substitution, and (2) potential modulated reduction of Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> NS to Bi NS. The first step governs the ultimate morphology and formation of defects, while the subsequent step determines the final composition and oxidation states.

### 3.5 Crystallinity

The amorphous photo/electrocatalysts benefit from abundant active sites, exotic electronic configuration, and flexible structure. These features expedite the adsorption of reaction mediators and facilitate photon/electron transfer between metal sites and intermediates, thereby promoting the transformation of precatalysts into active components.<sup>183,184</sup>

The impact of crystallinity on reconstruction, especially concerning amorphous structures, is multifaceted. Amorphous structures exhibit numerous influential characteristics: they possess abundant active sites and structural flexibility due to their lack of defined arrangement, enabling enhanced adsorption crucial for catalysis. Moreover, the surplus of dangling bonds within these structures significantly reduces the energy barrier required for rapid reconstruction into active phases, while the absence of dislocations and grain boundaries further facilitates this transformation process. Additionally, the tunable energy levels inherent in amorphous structures play a pivotal role in expediting the transition into active phases, thereby promoting efficient catalytic reactions. For example, the active phases with high energy evolved from amorphous photo/electrocatalysts improve the inherent catalytic efficacy of their active sites. Cai *et al.* prepared amorphous NiFe alloy catalysts for the OER by a solution method at room temperature.<sup>140</sup> An anodic potential was applied to activate the amorphous NiFe exposing more active sites due to the short-range order of the amorphous structure. The amorphization features of the synthesized NiFe alloy catalyst were confirmed by HAADF-STEM imaging (Fig. 16a) and the annealed NiFe alloy catalyst at 600 °C in nitrogen (Fig. 16b) was provided for comparison. Though both amorphous and crystalline catalysts show comparable electrochemically active surface area (Fig. 16c), the amorphous NiFe alloy achieved an overpotential of 265 mV ( $\eta_{10 \text{ mA cm}^{-2}}$ ), which is 100 mV lower than that of the

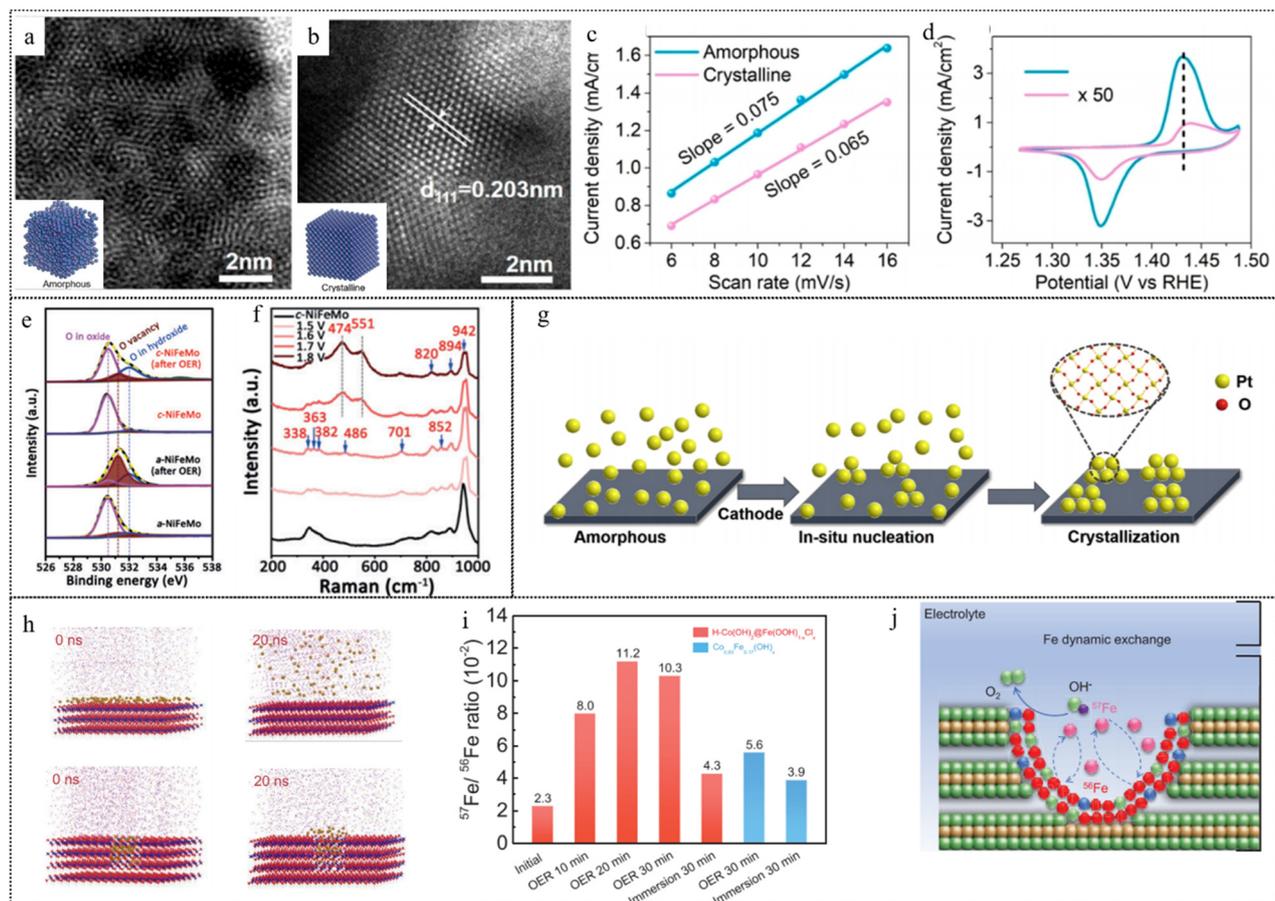
NiFe crystallized alloy. According to the CV measurements in Fig. 16d, the redox peaks of the amorphous catalyst are 50 times stronger than those of the crystalline form. The reconstruction of amorphous catalysts greatly increases the number of active sites and thus significantly improves the water oxidation activity.

Duan *et al.* reported a scalable production method of amorphous NiFeMo oxides, which can produce up to 515 grams of catalyst in one batch.<sup>141</sup> From the O 1s XPS spectra depicted in Fig. 16e, it can be observed that the concentration of O vacancies in the *a*-NiFeMo catalyst after the OER is greater than that in the *c*-NiFeMo catalyst. The rapid reconstruction of the amorphous structure was revealed by Raman spectroscopy (Fig. 16f) that the *a*-NiFeMo oxide reconstructed rapidly on the surface under OER conditions, generating a metal oxy(hydroxide) active layer with an enrichment of O vacancies, while the *c*-NiFeMo oxide shows no significant structural change. The authors also demonstrate by DFT calculations that O vacancies can lower the dissociation barrier of \*OOH intermediates, thus enhancing the OER activity.

In addition, transforming catalysts into an amorphous state is another effective strategy to enhance the precatalyst's contact area. The abundance of defects and disarrayed configurations resulting from amorphization allow for better electrolyte infiltration into the inner layer of precatalysts, thereby boosting the degree of reconstruction. This is a characteristic not typically observed in crystalline catalysts. For example, Chen *et al.* presented the utilization of fast reconstruction of amorphous nanofilm precatalysts for designing superior water-splitting catalysts.<sup>142</sup> The authors amorphized SrCo<sub>0.85</sub>Fe<sub>0.1</sub>P<sub>0.05</sub>O<sub>3- $\delta$</sub>  (SCFP) nanofilm precatalysts with weakened chemical bonds by using high-energy argon plasma to destroy the crystal structure of composite oxides, and deposited them on conductive nickel foam (NF) substrates. Under water-splitting operation conditions, the amorphous SCFP-NF catalysts could rapidly reconstruct within 2 minutes, forming novel amorphous phases with high activity and stability. The SCFP-NF catalysts after reconstructing activation exhibited ultrahigh mass activity of up to 1000 mA mg<sup>-1</sup> and excellent durability of up to 650 hours.

Amorphization structures typically exhibit unique and appealing catalytic performance, and the photo/electrochemical performance can be improved by *in situ* crystallization to obtain the optimized crystal structure. For example, Ma *et al.* developed a facile approach of the self-activation of amorphous platinum (A-Pt) nanospheres.<sup>143</sup> As illustrated in Fig. 16g, A-Pt was *in situ* reconstructed to a crystalline phase and partially oxidized during the CV activation cycling. This process led to the self-activated enhancements of both HER and HOR activities. As the electrochemical cycling process progresses, A-Pt gradually transforms into crystalline clusters and is subsequently oxidized. The formed Pt–O connection emerges as a key factor contributing to the elevated activities of the HER and HOR. Fine facet engineering could also facilitate the dynamic reconstruction of Fe ions on the surface, creating more active sites for the OER and thus prolonging the stability of the





**Fig. 16** HAADF-STEM images of (a) the as-prepared amorphous nickel-iron alloy catalyst and (b) the annealed nickel-iron alloy catalyst at 600 °C in nitrogen. (c) Double-layer capacitance acquired at a potential from 1.01 to 1.05 V vs. RHE to determine the ECSA. (d) CV curves obtained at a scan rate of 10 mV s<sup>-1</sup>. This figure has been reproduced from ref. 140 with permission from American Chemical Society, copyright 2020. (e) O 1s XPS spectra of a-NiFeMo and c-NiFeMo before and after the OER. (f) *Operando* Raman spectroscopy measurements of a-NiFeMo. This figure has been reproduced from ref. 141 with permission from Wiley-VCH, copyright 2019. (g) Schematic illustration of the *in situ* crystallization and activation of A-Pt during the electrochemical cycling process. This figure has been reproduced from ref. 143 with permission from American Chemical Society, copyright 2021. (h) Snapshots of Fe<sup>3+</sup> diffusion from the (0001) basal plane (upper) and a  $\beta$ -CoOOH nanohole (lower) in aqueous alkaline solutions at various times. (i) The <sup>57</sup>Fe/<sup>56</sup>Fe ratios of the initial H-Co(OH)<sub>2</sub>@Fe(OOH)<sub>1-x</sub>Cl<sub>x</sub> and the catalysts after chronopotentiometric OER testing for various times as well as soaking for 30 min in an electrolyte containing 1 ppm <sup>57</sup>Fe<sup>3+</sup>. (j) Schematic representation of the isotope-labeled Fe exchange mechanism in the OER. This figure has been reproduced from ref. 144 with permission from American Chemical Society, copyright 2023.

catalyst. Xue and coworkers used <sup>57</sup>Fe isotope labeled mass spectrometry (MS) to demonstrate that the catalyst achieved a local high-flux Fe dynamic equilibrium inside the nanoholes for the OER, which boosted the OER activity and reduced the Fe leaching amount.<sup>144</sup> The authors used DFT and molecular dynamics simulations (Fig. 16h) to indicate that the (10 $\bar{1}$ 0) facets on the nanohole walls exhibited stronger Fe adsorption than the (0001) basal plane. Additionally, the nanoholes were found to impede Fe diffusion from the catalyst to the electrolyte. The <sup>57</sup>Fe isotope labeled ICP-MS measurement of the OER mechanism was tested to investigate the dynamic Fe interchanging mechanism of the catalyst under the OER conditions. Isotope <sup>57</sup>Fe<sup>3+</sup> as the tracer and was added to the electrolyte. As shown in Fig. 16i, the <sup>57</sup>Fe content in the catalyst was about 5 times higher compared to that of the initial state after the OER process. Combining experimental and theoretical results, the proposed dynamic reconstruction is illustrated in Fig. 16j. The

free Fe species exhibited a distinctive preference for selective and robust adsorption onto the (10 $\bar{1}$ 0) sites situated along the walls of the nanoholes, as opposed to the (0001) basal plane. Simultaneously, the nanohole structure exerted a constraining effect, impeding the dissolution of Fe from the catalyst to the electrolyte (Fig. 17).

## 4. Applications

### 4.1 Electrocatalysis

Reconstruction in electrocatalysis can be attributed to surface energy drive or potential drive, the former causes topological changes, and the latter causes oxidation or reduction reactions. (Fig. 18) The pre-catalysts can be tuned in their structural reconstruction, composition changes, and surface properties to demonstrate significantly enhanced conductivity, accelerated charge



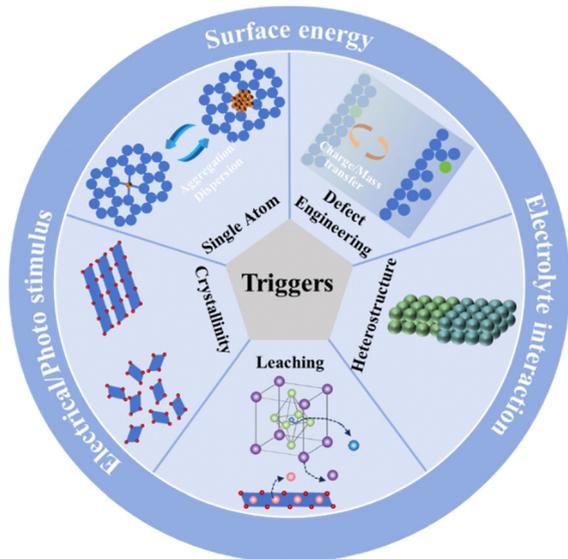


Fig. 17 Triggers and transformation mechanisms of catalyst reconstruction behavior.

transfer kinetics, and decreased energy barriers required for the formation of active intermediates. Reconstruction is applied in novel material synthesis and other electrocatalysis (such as the OER/ORR, UOR, MOR, HER, NORR and CO<sub>2</sub>RR), and several representative examples are provided.

In the realm of electrocatalysis, reconstruction is a critical phenomenon akin to surface modification strategies. It refers to the spontaneous changes in the surface structure of a catalyst, which can impact both its lifespan and activity. The reconstruction of electrocatalysts is basically attributed to the applied cathodic/anodic potentials and the robust interactions with reactants or intermediates. In this section, we will introduce the representative applications of favorable catalyst dynamic transformation in the field of electrocatalysis.

**4.1.1 Electrocatalytic oxidation reactions.** Redox reactions occurring on a catalyst's surface can lead to structural modifications within the catalyst. Consequently, the initial structural integrity of the catalyst may not remain consistent throughout the electrolysis process. This reconstruction phenomenon mainly involves changes in surface properties, composition, and structural configuration due to extended oxidation reactions.

**4.1.1.1 Oxygen evolution reaction (OER).** In the context of electrocatalytic water splitting, the OER is one of the half-reactions, and its kinetic process is typically sluggish. Therefore, the actual overpotential for the OER is often very high. In many previous studies, it has been found that high-valent transition metal species exhibit excellent OER catalytic activity. Taking the NiFe system as an example, some research suggests that high-valent Ni(IV) serves as the active site,<sup>185–187</sup> while others propose high-valent Fe(IV)<sup>188,189</sup> or even Fe(VI)<sup>190,191</sup> as the active site. However, one point universally agreed upon is that the generation of high-valence active components during dynamic transformation is crucial for enhancing OER activity.

Reconstruction during the OER process can be used not only to improve the catalytic activity of the catalyst but also to effectively design seawater electrolysis catalysts, improving the selectivity of the OER in the presence of Cl<sup>-</sup> as mentioned in the previous section.

**4.1.1.2 Urea oxidation reaction (UOR).** For the urea oxidation reaction (UOR), published works are focusing on developing the UOR process as a viable solution for wastewater treatment and converting it into energy. The UOR process has a thermodynamic potential of 0.37 V (vs. reversible hydrogen electrode, RHE), which is much lower than that of the energy-intensive oxygen evolution reaction (OER; 1.23 V vs. RHE). This makes the UOR process a promising candidate for replacing the OER process. Nickel hydroxides, acting as a typical and high-performance catalyst for the UOR, generate NiOOH and other active species by reconstruction.<sup>192</sup> However, the production of these reactive species requires a significant amount of energy. Therefore, researchers are focusing on finding ways to reduce the energy barriers associated with the reconstruction process. One approach is to modulate the electronic state or ligand structure of the active site, which can reduce the generation energy of reactive species such as NiOOH. This modulation promotes the reconstruction reaction and improves the catalytic activity of the UOR process. Cao *et al.* reported the synthesis and characterization of amorphous nickel sulfoselenide and revealed that nickel sulfoselenide is transformed into nickel oxyhydroxide during the UOR. The *in situ* formed nickel oxyhydroxide has a high reaction rate and selectivity for urea oxidation, leading to a lower cell voltage than that required for overall water splitting.<sup>193</sup>

**4.1.1.3 Methanol oxidation reaction (MOR).** Methanol oxidation reaction has been applied in direct methanol fuel cells since the 1950s for the convenience and safety of ethanol transportation.<sup>194</sup> Another application is that the MOR can be used to replace the OER for electrolyzers due to the lower thermodynamic equilibrium potential of 0.04 V vs. RHE. Though great progress has been made in the development of highly active Pt/Pd SACs for the MOR, further studies are needed to investigate the dynamic structural reconstruction of SACs under working conditions and break the scaling relation limit of DFT calculations,<sup>113,195</sup> since SACs may aggregate into clusters as true catalytic active species and then return to the initial atomic dispersion under ambient exposure. As for non-noble metal-based catalysts, the onset potential is high enough to oxidize catalysts during reactions. Zhang *et al.* reported the construction of a bismuth oxyhydroxide (BiO<sub>x</sub>(-OH)<sub>y</sub>)-Pt inverse interface *via* electrochemical reconstruction for the enhanced MOR.<sup>196</sup> The electrochemical treatment illustrated in Fig. 19a can induce the structural reconstructing of PtBi catalysts, leading to the formation of the BiO<sub>x</sub>(OH)<sub>y</sub>-Pt inverse interface.

**4.1.2 Electrocatalytic reduction reactions.** Cathodic reduction reactions are generally used to produce fuels or value-added chemicals at a lower potential than oxidation



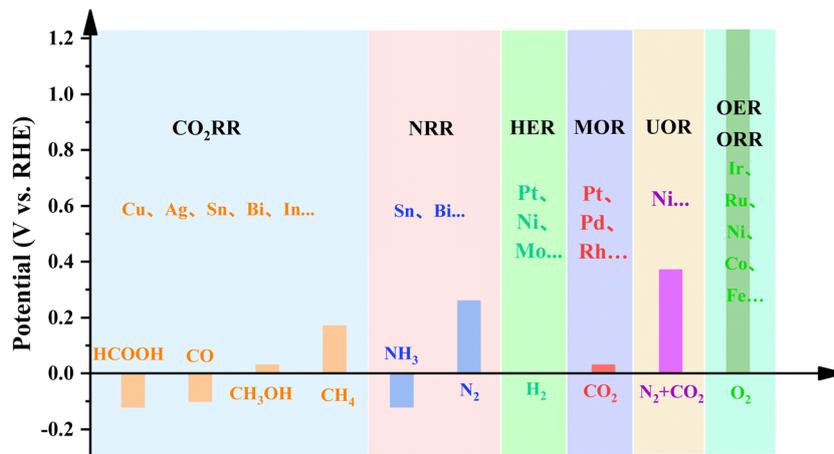


Fig. 18 Schematic diagram of electrocatalytic reactions and the related standard equilibrium potentials versus the reversible hydrogen electrode (RHE) under standard conditions. (Note: typical elements for specific electrocatalytic reaction and corresponding reconstruction are listed in elliptic boxes.)

reactions, which alters the valence state and surface morphology of electrocatalysts. These changes involve the reduction or oxidation of electrocatalysts depending on applied potentials, along with concurrent alterations in the composition and phase.

**4.1.2.1 Oxygen reduction reaction (ORR).** As for the ORR, reconstruction in transition metal-based catalysts often implies a loss of stability, such as being prone to oxidation, aggregation or dissolution in the electrolyte.<sup>200–202</sup> M–C–N (metal, metal-free) composites possess good qualities of low cost, high tolerance, high activity, and high stability.<sup>203,204</sup> In other words, an electrocatalyst for the ORR depends more on the structural stability to offer stable oxygen adsorption sites, where the reconstruction process is usually unfavorable and requires control.

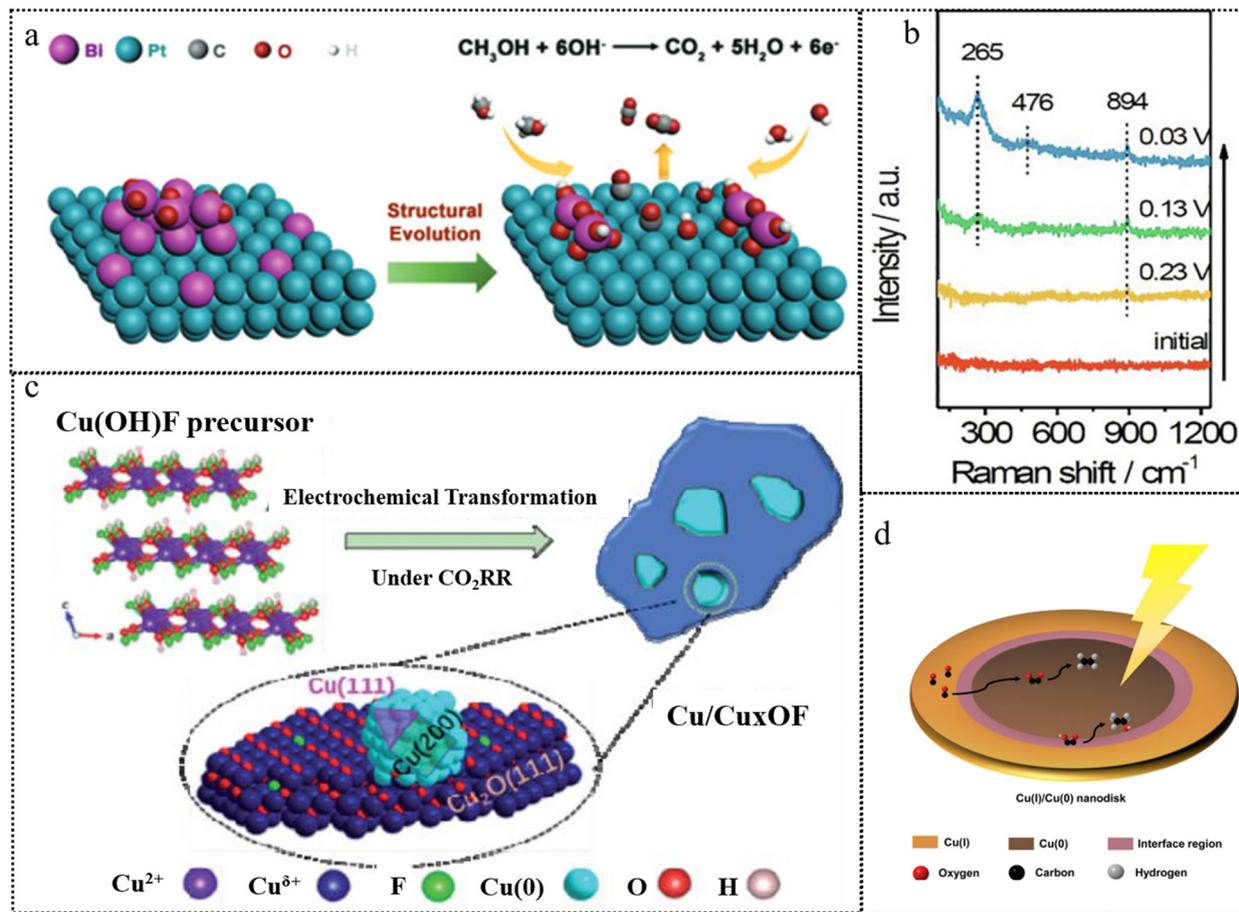
**4.1.2.2 Hydrogen evolution reaction (HER).** In the HER, both the reduction and oxidation processes of electrocatalysts are involved. For example, nickel sulfides stand out with high intrinsic activity and good electron transport capability.<sup>205</sup> The Ni element whether in the Ni(II) or Ni(III) form can undergo reduction at cathodes during the HER. Zheng *et al.* showed that two-dimensional NiS<sub>2</sub> was activated and transformed into nanosheets with metallic Ni.<sup>206</sup> In addition to reduction during the HER, the operation conditions affect the oxidation of electrocatalysts, as the redox potentials of the elements involved are partly overlapped by the applied potentials. For Mo-based catalysts, the potential required for the conversion of Mo to MoO<sub>4</sub><sup>2–</sup> is lower than that required for water splitting.<sup>207</sup> The dissolved MoO<sub>4</sub><sup>2–</sup> also plays a key role in electrocatalyst stability.<sup>197</sup> As further confirmed by *in situ* Raman spectroscopy (Fig. 19b), MoO<sub>4</sub><sup>2–</sup> was re-adsorbed and polymerized to Mo<sub>2</sub>O<sub>7</sub><sup>2–</sup> on the electrode surface, mitigating the strong \*H adsorption on Ni sites and enhancing the alkaline HER.

**4.1.2.3 Nitrogen reduction reaction (NRR).** Current electrocatalytic NRR processes are primarily centered on the production of NH<sub>3</sub> *via* the reduction reaction from N<sub>2</sub>, which is energy-

intensive for the inert nature of N<sub>2</sub>. Meanwhile, a significant quantity of nitrate (NO<sub>3</sub><sup>–</sup>), an alternative source for the production of NH<sub>3</sub>, is being released into surface water and underground aquifers, posing a risk to human health.<sup>208</sup> Zhang *et al.* synthesized CuO nanowire arrays and used <sup>15</sup>N isotope labeling confirming that the produced NH<sub>3</sub> originates from NO<sub>3</sub><sup>–</sup> reduction. *In situ* Raman and other *ex situ* analyses have demonstrated the electrochemical reduction from CuO to Cu/Cu<sub>2</sub>O during the electrocatalysis process, confirming the electron transfer at the Cu/Cu<sub>2</sub>O interface, which acts as the active phase.

**4.1.2.4 CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR).** The microscopic states of active sites affect the CO<sub>2</sub>RR, so determining the active intermediates and monitoring their changes under working conditions is pivotal.<sup>209,210</sup> Additionally, the CO<sub>2</sub>RR operates at more negative potentials, facilitating the reduction of Cu oxides or hydroxides *in situ*.<sup>211</sup> Reconstruction has been employed to move the reduction reaction towards well-constructed chemical compositions of the catalyst surface.<sup>212</sup> Halogen anions commonly exhibit robust interactions with copper atoms, substantially influencing the reconstruction processes of Cu. They are frequently employed to stabilize the highly active intermediate valence states of copper. Yang *et al.* used 2D Cu(OH)F nanoplates as a precursor and transformed them into Cu/Cu<sub>x</sub>OF by electrochemical treatment under the same conditions as those used for the alkaline CO<sub>2</sub>RR (Fig. 19c).<sup>198</sup> They revealed that the introduction of F not only modulated the exposed facets of Cu nanoparticles, but also stabilized the oxidized Cu species and induced the s,p-d coupling between the metal and heteroanions, which enhanced the electroactivity and selectivity for the acetate formation *via* the CO<sub>2</sub>RR. Yang *et al.* conducted another study, where they synthesized a range of P-Cu<sub>x</sub>/Cu<sub>2</sub>OF catalysts with varying Cu(I)/Cu(0) ratios.<sup>199</sup> This was achieved through the pulsed potential method with Cu(OH)F as the precursor. Characterization of the morphology and structure confirmed a nanodisk morphology comprising a Cu(0) core and a satellite-like Cu<sub>2</sub>OF shell. This arrangement provided exposed





**Fig. 19** (a) Schematic diagram of the preparation of  $\text{BiO}_x(\text{OH})_y\text{-Pt}$ . This figure has been reproduced from ref. 113 and 195 with permission from American Chemical Society, copyright 2020. (b) *In situ* Raman spectra of  $\text{Ni}_4\text{Mo}$  during the HER. This figure has been reproduced from ref. 197 with permission from The Royal Society of Chemistry, copyright 2019. (c) Schematic diagram of atomic structures of  $\text{Cu}(\text{OH})\text{F}$  and the transformed  $\text{Cu}/\text{Cu}_x\text{O}$ . This figure has been reproduced from ref. 198 with permission from The Royal Society of Chemistry, copyright 2021. (d) Schematic diagram of the  $\text{Cu}(\text{I})/\text{Cu}(\text{O})$  core/shell nano-disk for  $\text{P-Cu}_{1.65}/\text{Cu}_2\text{O}$ . This figure has been reproduced from ref. 199 with permission from Elsevier, copyright 2023.

sites of  $\text{Cu}(\text{I})$ ,  $\text{Cu}(\text{O})$ , and the  $\text{Cu}(\text{I})/\text{Cu}(\text{O})$  interface (depicted in Fig. 19d). The interfacial region was found favorable to ethanol formation, yielding a high  $\text{FE}(\text{C}_{2+})$  of up to 80.2% and achieving a notable partial current density of up to  $128 \text{ mA cm}^{-2}$ .

**4.1.3 Stability of the OER in practical electrolyzers.** Although most of the OER activity and stability studies were carried out on standard, lab-scale three-electrode devices, this does not necessarily directly reflect its performance in practical electrolyzers.<sup>213,214</sup> It may not be possible to fully simulate the actual operating conditions in laboratory tests, e.g. the formation of microbubbles on the surface of the electrode in rotating disc electrode (RDE) experiments may lead to misleading stability test results. In a practical electrolyzer, the operating conditions are usually much more demanding and there are many more variables that can affect the stability behaviors of the catalyst.<sup>215</sup> Therefore, in order to ensure that the catalysts developed can maintain activity and stability in more complex practical application environments, and to apply these catalysts to commercial-scale electrolysis, a bridge needs to be built between the lab-scale test system and the practical electrolyzer.

Firstly, the preferred surface reconstruction product of the catalyst may be different in commercial-scale operation, typically at high current density and voltage. In addition, the oxygen evolution pathway could be altered by possible morphological or electronic structure changes on the catalysts in the electrolyzer. Membranes and ionomers, which can affect the local pH environment, as well as different catalyst supports, can all affect the charge density distribution on the catalysts. Moreover, the dissolution-redeposition process may have a more significant effect in batch or recyclable electrolyte systems; however, in a flow electrolyzer the dissolved metal ion concentration may be lower, reducing the influence of the ions.

Currently, water electrolysis technology involves conventional alkaline water electrolyzer (AWE) as well as two emerging promising designs: a proton exchange membrane water electrolyzer (PEMWE) and an anion exchange membrane water electrolyzer (AEMWE) (Fig. 20a–c). The latter two can significantly reduce energy costs due to the compact configuration of their membrane electrode assemblies, low ionic resistance and reduced ohmic drop.<sup>216,217</sup> The anodic reaction of PEMWE and



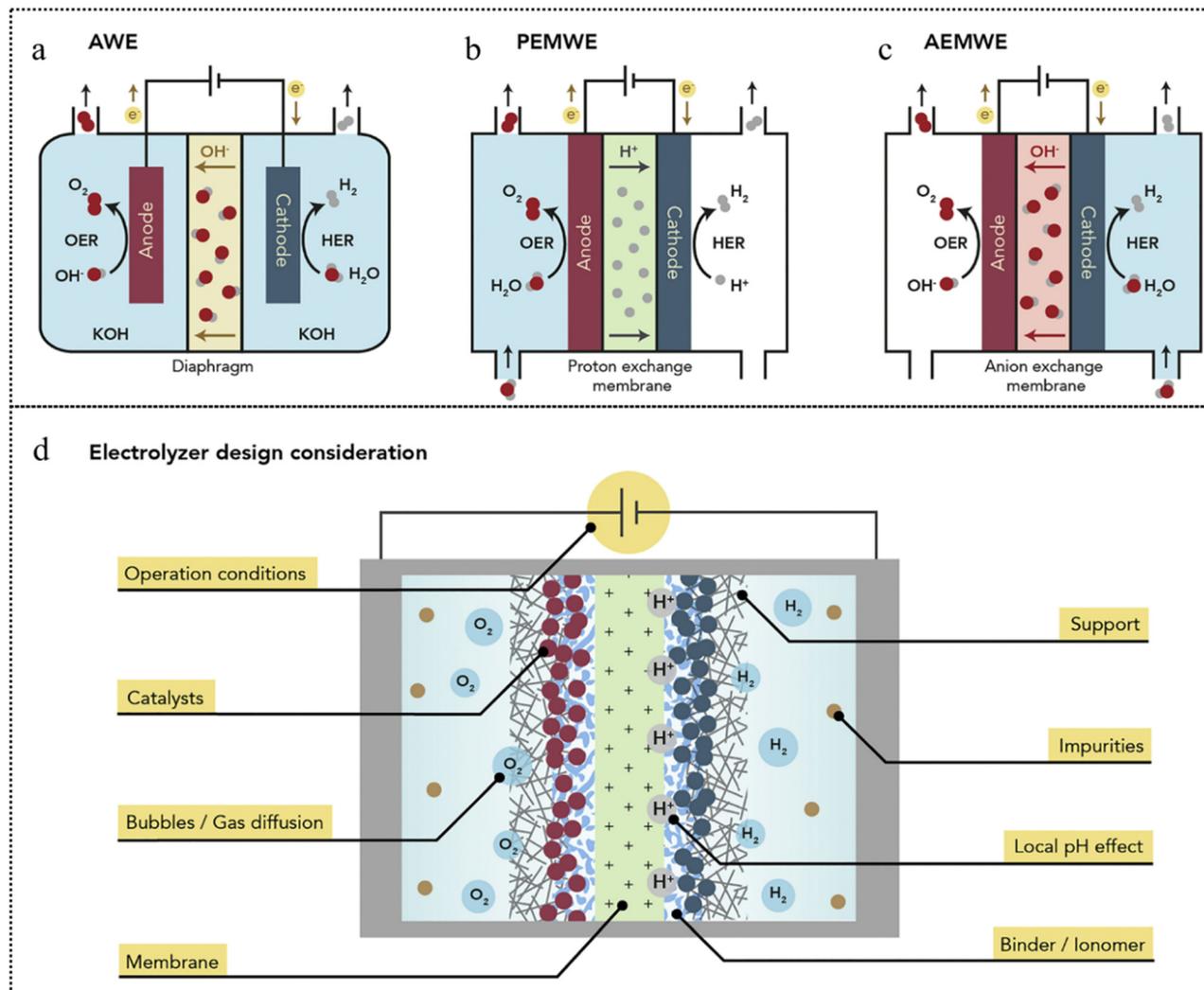


Fig. 20 (a)–(c) Leading configurations of current water electrolyzers and design considerations for long-term stability, (d) Critical elements that need to be considered for fabricating electrolyzers with long-term stability. This figure has been reproduced from ref. 219 with permission from Elsevier, copyright 2021.

AEMWE is affected by membrane properties and local pH, with the oxygen evolution reaction in acidic and alkaline environments, respectively. The PEMWE technology is well established and widely used, but faces the challenge of developing a non-precious metal OER catalyst. AEMWE, on the other hand, has advantages due to the applicability of a wider range of catalysts, nevertheless, membrane stability and electrolyzer design need to be improved to meet long-term operational requirements.<sup>218</sup> These key factors in design such as membrane materials, binders, supports (Fig. 20d), *etc.* will be explored in further detail.

**4.1.3.1 Membranes and ionomers.** Membrane stability is critical in electrolyzer design. Proton exchange membranes have been developed to commercial levels of durability, with Nafion membranes being widely used in PEMWE equipment, although membrane thinning and degradation may occur at high temperatures and current densities.<sup>220</sup> This may be due to the

formation of peroxides and free radicals caused by the crossing of oxygen on the cathode side, leading to membrane degradation under certain operating conditions. While anion exchange membranes have been improved to degrade only slightly over thousands of hours, improvements are still needed to create durable AEMWE devices.<sup>221</sup> Improving the conductivity of hydroxide ions in anion exchange membranes is a challenge in long-term operation. The study proposes a new method for preparing MEA that enhances conductivity by creating an additional  $\text{OH}^-$  transport pathway.<sup>222</sup> Factors such as degradation of polymer functional groups, blockage of ion exchange pathways, and the effect of temperature and electrolyte on membrane stability need to be investigated in order to enhance membrane durability.

Ionomers are a key factor in electrolytic stability, acting as a binder and ionic conductor for the electrode catalyst. Their content needs to be optimized for the catalyst and operating conditions to avoid high ionic resistance or electrode



clogging.<sup>223</sup> The design of AEMWE's durable ionic polymers is more challenging than that of stable Nafion.

**4.1.3.2 Electrodes and support materials.** The selection of suitable support materials is critical for electrolytic stability. Studies have shown that the interaction between the catalyst and the support material affects the electronic structure and catalytic stability.<sup>224</sup> For example, the acidic OER behaviors of IrOx nanoparticles on carbon black and antimony-doped tin oxide varies depending on the support. Different support materials (e.g., glassy carbon, gold, and boron-doped diamond) also lead to different Ir stability. The NiFe oxyhydroxide system exhibits a similar phenomenon under alkaline conditions.<sup>225,226</sup> Differences in electronic structure and catalyst-support material interactions explain the importance of optimizing catalyst loading. Carbon-based support or gas diffusion layers may initiate carbon corrosion or oxidation, affecting catalytic stability, which is important for both practical electrolysis installations and laboratory-scale experiments.<sup>227</sup> In addition, regeneration of catalytic electrodes and decomposition of impurities on electrodes on commercial electrolyzers can be achieved by high-temperature pulse annealing.<sup>228</sup> These findings aid in the fabrication of durable OER electrolysis electrodes.

**4.1.3.3 Operating conditions and local environments.** When applying the results of laboratory-scale catalyst testing to commercial electrolyzers, it is important to consider the actual operating conditions. Studies have shown that the corrosion mechanisms of catalysts are related to voltage and current density, resulting in potentially different stability behavior in practical applications. Therefore, even if certain catalysts are stable in laboratory tests, they may not be able to withstand the high current densities and yield rates required by industry.<sup>229</sup> In addition, there may be differences in local pH between laboratory three-electrode cells and MEA electrolyzers, the latter of which may develop extreme pH environments due to factors such as the ion transport properties of the membrane, reactant diffusion and product surface coverage.<sup>230</sup>

At high current densities, the bubbles generated by the OER may differ in physical properties from those under laboratory conditions, leading to problems such as interfacial supersaturation, high catalytic surface coverage, and increased ohmic internal resistance, all of which may contribute to operational instability.<sup>231</sup> *In situ* TEM observations have shown that bubble formation and collapse can lead to oscillations in the catalyst structure due to water uptake and precipitation of oxygen within the oxide.<sup>232</sup> Therefore, care should be taken to monitor the operating conditions of the electrolyzer at high potentials and current densities to avoid catalytic instability.

**4.1.3.4 Impurities.** Electrolysis conditions in practical applications often differ from laboratory test environments, which typically use pure electrolytes for catalytic performance measurements, whereas in practice impure water sources and brines may be involved. This requires consideration of issues such as OER selectivity, catalytic site blockage and ion transport

pathway blockage.<sup>233</sup> NiFe-(oxy)hydroxide-based catalysts perform well in seawater decomposition and are suitable for use in alkaline AEMWE, but impurity deposition remains a challenge.<sup>234</sup> Electrolytes, impurities, and factors such as membranes, ionomers, support materials, and electrodes need to be considered when designing actual electrolyzers for long-term operation.

**4.1.3.5 Catalyst reconstruction in electrolyzers.** Significant breakthroughs have been achieved in the study of electrolyzers under industrial conditions. Developing non-precious metal anode catalysts for PEM electrolyzers is both attractive and highly challenging. Non-precious metals such as Fe, Co, and Ni, which perform excellently in alkaline environments, tend to undergo irreversible dissolution. However, recent research by Francisco Pelayo and colleagues has shown great breakthrough using cobalt tungstate (CWO) treated in an alkaline environment to achieve efficient and stable PEMWE catalysts.<sup>231</sup> The study demonstrated that the stable network of oxides and water hydroxides in CWO can achieve low overpotentials and high current densities. Under industrial conditions (80 °C), the CWO catalyst achieved a current density of 1.8 A cm<sup>-2</sup> at 2 V and maintained stability at 1.77 V for 600 hours, showcasing significant performance improvements. These findings highlight the potential of non-precious metal catalysts for PEM electrolyzers, paving the way for more cost-effective and sustainable solutions in industrial applications.

As for commercial alkaline water electrolyzers, the systems typically operate at 80 °C to minimize energy consumption. However, one of the most promising anode candidates, nickel-iron alloy catalysts, faces a high solubility bottleneck at this temperature. Sun and co-authors found that the dissolution of NiFe-LDH during operation not only leads to the degradation of the anode itself but also deactivates the cathode, resulting in an overall decline in electrocatalytic performance.<sup>229</sup> To mitigate this dissolution, oxygen anions were used as inhibitors in the electrolyte. Adding phosphates to the electrolyte was found to inhibit the loss of NiFe-LDH active sites at 400 mA cm<sup>-2</sup>, reducing the loss to one-third of its original level. This intervention decreased the performance degradation rate by 25 times. These findings underscore the importance of addressing the dissolution issue in high-temperature industrial applications, emphasizing the need for further research into effective inhibitors and stabilization strategies to enhance the longevity and performance of non-precious metal catalysts in commercial electrolyzers.

Other challenges remain in industrial applications. For example, in PEM systems, the quality of water required is very high. Impurities can lead to destructive catalyst reconstruction or degradation of the ion-exchange membrane, presenting critical challenges. Scholars from Graz University of Technology have revealed the fatal impact of chloride ion impurities in water on PEMWE under industrial conditions.<sup>224</sup> They found that even a chloride ion concentration as low as 10 ppm can significantly affect the performance of the electrolyzer. After 50 hours of operation, the low-frequency resistance of the

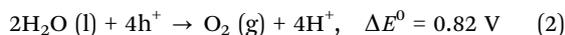


system increased from  $0.23 \Omega \text{ cm}^2$  by 100% at  $1.0 \text{ A cm}^{-2}$ . Additionally, they observed catalyst detachment and ion-exchange membrane degradation. This finding compels us to carefully consider the impact of using tap water, which contains residual chlorine, on the lifespan of PEM systems. While further purification of water could mitigate these effects, it would also increase operational costs. Therefore, balancing water quality and cost is a crucial issue that needs to be addressed for the sustainable operation of PEM systems.

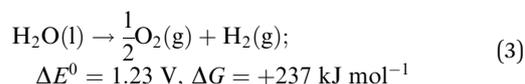
## 4.2 Photocatalysis

Photocatalysis represents a sustainable technology aimed at mitigating the global energy crisis and environmental pollution through the utilization of abundant and renewable solar energy resources. Typically, photocatalytic reactions entail the generation of charges upon exposure to light, subsequent charge separation and transportation, and ultimate charge consumption through surface redox reactions. In certain cases, alongside the use of semiconductor photocatalysts, cocatalysts are required to suppress charge recombination and facilitate charge transport efficiently. It is inevitable that charge carriers and reactants interact with the semiconductor and/or cocatalysts upon contact, resulting in the dynamic transformation of catalysts during the photocatalytic reaction. This dynamic transformation not only gives rise to real active sites, resembling those in electrocatalysis, but also provides an avenue for precisely fine-tuning the energy band structure of photocatalysts. In this section, we will introduce the representative applications of favorable catalyst dynamic transformation in the field of photocatalysis.

**4.2.1 Photocatalytic water splitting.** Hydrogen is increasingly regarded as a promising contender for replacing traditional fossil fuels due to its environmental benefits and high energy density. Photocatalytic water splitting stands out as a renewable and sustainable technology, directly harnessing inexhaustible water resources and solar energy to produce green hydrogen. As demonstrated in the following equation, during photocatalytic water splitting, photoinduced electrons act as reducing agents, while photoinduced holes serve as oxidants, yielding  $\text{H}_2$  and  $\text{O}_2$ , respectively.



Overall reaction:



**4.2.1.1 Co-catalysts.** Photocatalytic water splitting is inherently an uphill reaction. To facilitate this process, the band structures of photocatalysts must adhere to specific criteria, including a conduction band (CB) minimum that is more negative than the  $\text{H}^+/\text{H}_2$  reduction potential and a valence band (VB) maximum that is more positive than the  $\text{O}_2/\text{H}_2\text{O}$  oxidation

potential. Certainly, meeting all these stringent requirements with a single photocatalyst poses a significant challenge, making it necessary to introduce cocatalysts to enhance the charge separation efficiency and reaction kinetics. In general, electrocatalysts exhibiting high efficiency in electrocatalytic water splitting, such as Pt,<sup>85</sup> metal oxides, phosphides,<sup>235</sup> sulfides, carbides, and others, have strong potential as cocatalysts for photocatalytic water splitting.<sup>236</sup> These cocatalysts undergo dynamic transformations during the photocatalytic water-splitting process (Table 3).

Pt is frequently employed as a cocatalyst in the photocatalytic HER, and the beneficial impact of achieving a high dispersion of Pt in photocatalytic HER has been well-documented. However, these highly dispersed metal-based catalysts are susceptible to photochemical processes. During photocatalytic *operando* XAS experiments conducted on Pt single-atom catalysts (SACs), the typical trend observed in relation to the initial conditions is a reduction in Pt–O coordination, with coordination numbers decreasing from 4–5 to 2–3.<sup>237</sup> This indicates a partial reduction of Pt species, and the SACs exhibited partial Pt–Pt clustering under the reaction conditions. The anatase-rich P90 titania support could lead to extensive clustering with a Pt–Pt coordination number of 7, and this was significantly suppressed by employing a precalcined Pt/PC500 anatase-rich support with high-specific surface areas and high surface defectiveness. Specifically, Pt/PC500 maintained an atomic Pt dispersion and a high Pt oxidation state consisting of single atoms, dimers, and trimers, with Pt–Pt coordination numbers of 1–2 during the photocatalytic HER. Notably, using the same support, a mild reducing pretreatment can yield near-neutral and highly dispersed Pt species with Pt–O and Pt–Ti bonds in the pre-reduced Pt/PC500, leading to improved photocatalytic HER performance and stability ( $0.14 \text{ mol h}^{-1} \text{ g}^{-1}$ , a turnover frequency per Pt atom of  $0.37 \text{ s}^{-1}$ , a photonic yield of 5.6%, and an apparent quantum yield of 11%). In addition to the photocatalytic HER, Pt-based co-catalysts can be employed with great efficiency in photocatalytic overall water splitting. Taking a PtO/TiO<sub>2</sub> photocatalyst as an example, the Pt–O bond length extended from 2.07 to 2.13 Å, while the coordination number decreased from 4.0 to 2.5 during the photocatalytic overall water splitting.<sup>63</sup> The dynamic transformation of the PtO cocatalyst is instrumental in unilaterally quelling undesirable hydrogen oxidation and achieving exceptional efficiency in the hydrogen evolution rate.

Transition metal compounds have been proven as efficient cocatalysts for photocatalytic water splitting when integrated with appropriate semiconductors. When NiSe<sub>2</sub> with different phases were incorporated into carbon nitride (CN) to form a hybrid photocatalyst (*m*-NiSe<sub>2</sub>/CN and *p*-NiSe<sub>2</sub>/CN), these hybrids exhibited similar photocatalytic HER rates of  $3.26 \mu\text{mol h}^{-1}$  and  $3.75 \mu\text{mol h}^{-1}$ , respectively.<sup>236</sup> Strikingly, distinct surface conversion behaviors were observed in *m*-NiSe<sub>2</sub>/CN and *p*-NiSe<sub>2</sub>/CN during the photocatalytic process. After the photocatalytic HER, the chemical structure and morphology of *m*-NiSe<sub>2</sub> were well retained, but partial *p*-NiSe<sub>2</sub> would undergo conversion to NiOOH, resulting in different stability of





Table 3 Dynamic transformation in different photocatalytic applications and the corresponding catalytic performance

Catalysts	Dynamic transformation type	Application	Catalytic performance	Ref.
Pt/P90-C	Chemical state and coordination environment	HER	1.4 $\mu\text{mol min}^{-1}$	237
Pt/PC500-C	Chemical state and coordination environment	HER	1.6 $\mu\text{mol min}^{-1}$	237
Pt/PC500-R	Chemical state and coordination environment	HER	2 $\mu\text{mol min}^{-1}$ , AQY 11%	237
PtO/TiO <sub>2</sub>	Coordination environment	Overall water splitting	—	63
<i>p</i> -NiSe <sub>2</sub> /CN	Phase transformation	HER	3.75 $\mu\text{mol h}^{-1}$ ,	238
Ni@NiO <sub>x</sub> -SrTiO <sub>3</sub>	Phase transformation and morphology	Overall water splitting	H <sub>2</sub> yield: 0.3 $\mu\text{mol min}^{-1} \text{g}^{-1}$ O <sub>2</sub> yield: 0.1 $\mu\text{mol min}^{-1} \text{g}^{-1}$ 323.2 $\mu\text{mol h}^{-1}$	67
ME-TiO <sub>2</sub> @Ru	Chemical state	HER	15 $\mu\text{mol h}^{-1}$	239
Bi <sub>4</sub> TaO <sub>8</sub> Cl	Phase transformation	OER	20 $\mu\text{mol h}^{-1}$	240
Bi <sub>4</sub> TaO <sub>8</sub> Br	Phase transformation	OER	19.3 $\text{mmol h}^{-1} \text{g}^{-1}$	241
CdS/a-MoO <sub>3</sub> -32.6 wt%	Chemical state	HER	0.96 $\text{mmol g}^{-1} \text{h}^{-1}$	242
TP-COFs-1	Coordination environment	HER	56 $\mu\text{mol h}^{-1}$	243
QDs-Ni(OH) <sub>2</sub>	Phase transformation	HER	CO yield: 95.41 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR (Light irradiation: 200 $\text{mW cm}^{-2}$ )	CO yield: 109.89 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR (Light irradiation: 500 $\text{mW cm}^{-2}$ )	C <sub>2</sub> H <sub>4</sub> yield: 29.84 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR (Light irradiation: 1000 $\text{mW cm}^{-2}$ )	CO yield: 224.65 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	C <sub>2</sub> H <sub>4</sub> yield: 30.12 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 25.54 $\mu\text{mol g}^{-1} \text{h}^{-1}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CO yield: 92.0 $\mu\text{mol g}^{-1} \text{h}^{-1}$	219
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 11.3 $\mu\text{mol g}^{-1} \text{h}^{-1}$	245
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CO yield: 12.3 $\mu\text{mol g}^{-1} \text{h}^{-1}$	245
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	C <sub>2</sub> H <sub>4</sub> yield: 18.5 $\mu\text{mol g}^{-1} \text{h}^{-1}$	245
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 11.6 $\mu\text{mol g}^{-1} \text{h}^{-1}$	245
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CO yield: $\sim$ 21.36 $\mu\text{mol g}^{-1} \text{h}^{-1}$	246
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CO yield: 0.0924 $\text{mmol}$	244
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 241.6 $\mu\text{mol g}^{-1} \text{h}^{-1}$	247
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 1.19 $\mu\text{mol g}^{-1} \text{h}^{-1}$	248
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 0.41 $\mu\text{mol g}^{-1} \text{h}^{-1}$	248
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 0.87 $\mu\text{mol g}^{-1} \text{h}^{-1}$	248
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	CO <sub>2</sub> RR	CH <sub>4</sub> yield: 168.5 $\mu\text{mol g}^{-1} \text{h}^{-1}$	249
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	Methanol reforming reaction	H <sub>2</sub> yield: 0.64 $\text{mmol g}^{-1} \text{h}^{-1}$	250
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	Ch(vi) removal	17.14 $\text{mg L}^{-1} \text{h}^{-1}$	251
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	Phenol degradation	8 $\text{ppm h}^{-1}$	252
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	Phenol degradation	5.3 $\text{ppm h}^{-1}$	252
Pt <sub>1</sub> /ZnNiTi-LDHs-E	Chemical state	Phenol degradation	9 $\text{ppm h}^{-1}$	253

*m*-NiSe<sub>2</sub>/CN and *p*-NiSe<sub>2</sub>/CN. Specifically, *m*-NiSe<sub>2</sub>/CN maintained stable H<sub>2</sub> evolution, whereas *p*-NiSe<sub>2</sub>/CN experienced a significant 57.1% reduction in catalytic activity over 25 hours.

In addition, core-shell co-catalysts consisting of Ni@NiO<sub>x</sub> were employed to functionalize SrTiO<sub>3</sub> (BSTO-1000-NiO<sub>x</sub>) for photocatalytic overall water splitting.<sup>67</sup> Under the influence of humidity and aqueous conditions, the NiO phase underwent transformation into Ni(OH)<sub>2</sub>. In the early stages of illumination, excess hydrogen production with an H<sub>2</sub>:O<sub>2</sub> yield ratio exceeding 2 implied the oxidation of Ni(OH)<sub>2</sub> to NiOOH, serving as a cocatalyst for water oxidation. Upon discontinuation of illumination, a disproportionation reaction took place involving Ni and NiOOH, giving rise to the formation of Ni(OH)<sub>2</sub> with residual embedded Ni. The subsequent reinitiation of illumination elucidated the recurring transients noted in the formation of excess hydrogen and oxygen (Fig. 21a).

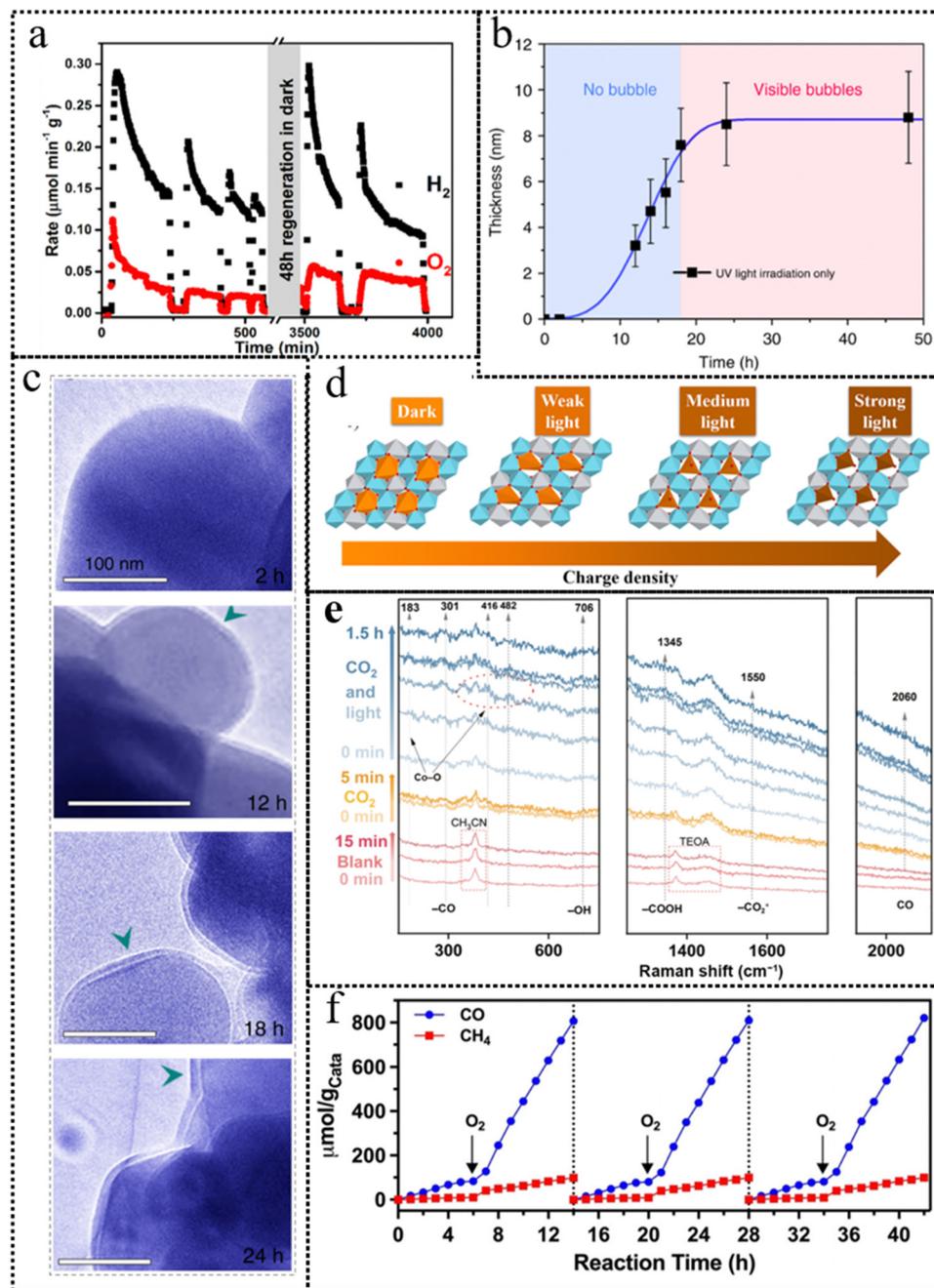
**4.2.1.2 Semiconductors.** As one of the most significant semiconductor photocatalysts, TiO<sub>2</sub> has garnered substantial interest and extensive research over several decades. Recently, the dynamic transformation of anatase TiO<sub>2</sub> nanoparticles in photocatalytic H<sub>2</sub> evolution was investigated by using liquid environmental TEM. As depicted in Fig. 21b and c, a self-hydrogenated shell was observed on the TiO<sub>2</sub> surface prior to the emergence of hydrogen bubbles. This shell was established through subsurface diffusion of H atoms created *via* the reaction of water protons with photoinduced electrons, playing a critical role in promoting hydrogen generation by lowering the activation energy for H<sub>2</sub> (H-H bond) formation.<sup>254</sup> Similar dynamic transformation phenomena were also observed on the Bi<sub>4</sub>TaO<sub>8</sub>X photocatalyst.<sup>240</sup> During the photocatalytic oxygen evolution process, photoinduced TaO<sub>x</sub> was *in situ* generated on the surface of the Bi<sub>4</sub>TaO<sub>8</sub>X photocatalyst at the start of irradiation and then underwent oxidation by the electron acceptor. The photoinduced TaO<sub>x</sub> could readily capture and react with hole scavengers or electron acceptors by virtue of the metastable chemical valence in an amorphous state, allowing photogenerated electrons and holes to participate in the photocatalytic HER and OER, respectively. As a result, the dynamic transformation process could not only yield active species for surface catalytic reactions but also enhance the separation of photogenerated charges, ultimately improving the overall efficiency of the photocatalytic reaction. However, dynamic transformation in metal oxide photocatalysts is not always beneficial to the reactions, such as the photocorrosion of copper oxide during the photocatalytic water splitting.<sup>257</sup> The CB potential of Cu<sub>2</sub>O promoted the rapid transfer of photoinduced electrons for H<sub>2</sub> evolution. However, photoinduced holes tended to accumulate and were responsible for severe Cu<sub>2</sub>O photocorrosion. This was caused by the relatively low VB energy level of Cu<sub>2</sub>O, which restricted the participation of photoinduced holes in water oxidation. The inability to efficiently remove photoinduced holes by adsorbed water molecules resulted in the self-oxidation of Cu<sub>2</sub>O. Introducing a hole scavenger with an appropriate oxidizing potential facilitated the transfer of photoinduced holes, thus enhancing Cu<sub>2</sub>O photostability.

Creating heterojunctions at the nanoscale is indeed an effective approach for improving the separation of photoinduced carriers, broadening the light absorption range, and facilitating mass transfer, all of which contribute to enhanced photocatalytic performance.<sup>258</sup> For heterojunction photocatalysts, their dynamic transformation may be affected by interactions between components.<sup>259</sup> In CdS/*a*-MoO<sub>x</sub> heterojunction photocatalysts, the initial photoexcited electrons in CdS demonstrated a robust reduction capacity and effectively transferred to *a*-MoO<sub>x</sub>, leading to the reduction of Mo<sup>6+</sup> to Mo<sup>5+</sup>.<sup>241</sup> This transformation initiated an *in situ* electronic structure reconstruction, and it was empirically demonstrated that this electronic reconstruction played a pivotal role in the progressive enhancement of photocatalytic H<sub>2</sub> evolution performance. For the CdS/*a*-MoO<sub>x</sub>-32.6 wt% heterojunction catalyst, there was a substantial 16.8-fold increase in the rates of hydrogen production during the progression of the photocatalytic HER, elevating the rate from 1.1 mmol h<sup>-1</sup> g<sup>-1</sup> to 19.3 mmol h<sup>-1</sup> g<sup>-1</sup>. It is noteworthy that this rate significantly outperformed the hydrogen evolution rate achieved by the CdS nanorod catalyst, which stood at 1.1 mmol g<sup>-1</sup> h<sup>-1</sup>.

Covalent organic frameworks (COFs) composed of C, H, O, N, S and halogen elements represent a novel class of semiconducting polymeric materials for photocatalysis, which are gaining prominence by virtue of their exceptional attributes, including stable ordered crystal structures, high porosity and tunable light absorption, as well as versatile 2D or 3D network topologies. Very recently, the enol-to-keto dynamic transformation in the BT-COFs with imine linkages during the photocatalytic H<sub>2</sub> evolution was studied.<sup>242</sup> The enol-to-keto conversion in BT-COFs-1 was found to be thermodynamically favorable but kinetically sluggish, leading to the predominant presence of the enol form. Alkaline post-treatment effectively promoted dynamic transformation, leading to an increased population of keto-injectors within the structural building blocks, which could improve photocatalytic performance. The generation of keto-injectors not only rendered the CB level more negative but also introduced non-uniform charge distribution within the donor-acceptor molecular building blocks, thereby creating a significant intramolecular built-in electric field. Benefiting from the aforementioned advantages, TP-COFs-1, featuring a single keto group within its skeletal building blocks, exhibited excellent photocatalytic performance, yielding a remarkable H<sub>2</sub> evolution rate of 0.96 mmol g<sup>-1</sup> h<sup>-1</sup>.

**4.2.2 Photocatalytic CO<sub>2</sub> reduction reactions.** The escalating levels of CO<sub>2</sub> emissions present a formidable challenge to human existence and sustainable development.<sup>260,261</sup> Photocatalytic CO<sub>2</sub>RR is a promising approach to address both the energy crisis and the greenhouse effect by converting CO<sub>2</sub> into valuable fuels and chemical feedstocks. Photocatalytic CO<sub>2</sub>RR primarily yields products in two categories: (1) carbonates generated by the fixation of CO<sub>2</sub> into epoxides, and (2) valuable chemicals such as CH<sub>3</sub>OH and CO.<sup>242</sup> CO<sub>2</sub> activation (CO<sub>2</sub> → CO<sub>2</sub><sup>-</sup>) is a challenging process as it requires a higher potential compared to the conversion of CO<sub>2</sub> into fuels and chemicals. Designing an ideal CO<sub>2</sub>RR catalyst is a complex task due to the





**Fig. 21** (a) Transient behavior in  $\text{H}_2$  and  $\text{O}_2$  evolution by using 25 mg of BSTO-1000- $\text{NiO}_x$  (black:  $\text{H}_2$ ; red:  $\text{O}_2$ ). This figure has been reproduced from ref. 67 with permission from American Chemical Society, copyright 2020. (b) Thickness of the surface shell on anatase  $\text{TiO}_2$  vs. UV illumination time: experimental data (black squares) and fitted curve based on the KJMA equation 21 (blue line). Growth of the surface shell experienced a sudden decrease when bubbles became visible at around 18 hours (c) Photocatalysis experiments with different times of UV exposure. Each TEM image was recorded independently with fresh aqueous suspensions of  $\text{TiO}_2$  NPs controlled by the fluidic holder. This figure has been reproduced from ref. 254 with permission from Nature, copyright 2018. (d) Illustration of the dynamic transformation of Pt SAs under light irradiation. This figure has been reproduced from ref. 244 with permission from American Chemical Society, copyright 2022. (e) Pre-*operando* Raman spectra collected during the photocatalytic  $\text{CO}_2$  reduction. This figure has been reproduced from ref. 255 with permission from Springer, copyright 2023. (f) In the cyclic  $\text{CO}_2$  photoreduction experiment on pre-treated Cu-ZnTCPP/g- $\text{C}_3\text{N}_4$ , the gas atmosphere was alternated between aerobic and anaerobic conditions in each cycle. This figure has been reproduced from ref. 256 with permission from Wiley-VCH, copyright 2023.

diversity of products and multielectron processes, necessitating both high photocatalytic activity and selectivity for the desired product. Furthermore, photocatalytic  $\text{CO}_2$  reduction faces significant challenges of low  $\text{CO}_2$  affinity and competition from the HER.

To explore the dynamic transformation of single atom catalysts during the photocatalytic  $\text{CO}_2\text{RR}$ , metal ion vacancy-rich ZnNiTi layered double hydroxide-etched (ZnNiTi-LDHs-E) supports were utilized to immobilize Pt single atoms (Pt



SAs).<sup>244</sup> As illustrated in Fig. 21d, the Pt SAs exhibited distinct photochemical responses, forming cationic Pt(IV), electron-rich Pt(II), and near-neutral Pt<sup>δ+</sup> species under varying light intensities. The *in situ* generation of the various Pt SAs described above significantly influenced the product distribution in photocatalytic CO<sub>2</sub> reduction. Under weak light conditions, Pt(IV) SAs failed to activate CO, thereby preventing further transformation into hydrocarbons. Moderate light exposure resulted in electron-rich Pt(II) SAs, which could collaborate with adjacent LDH surface sites (Ni<sup>2+</sup>/Ti<sup>4+</sup>) to enable the C–C coupling route, leading to C<sub>2</sub>H<sub>6</sub> generation. The near-neutral Pt<sup>δ+</sup> species play a crucial role in the direct hydrogenation of C1 to generate CH<sub>4</sub>. This research revealed the dynamic transformation of Pt SAs and deepened the comprehension of active site properties in artificial photosynthesis, offering valuable insights for the development of efficient SA photocatalysis.

The surface coordination structure of transition metal-based semiconductors is prone to alteration upon exposure to environmental O<sub>2</sub> due to the coordinated nature of their 3d-orbital electrons. For example, the *in situ* formation of CoO<sub>x</sub> active sites on the Co (CoP) surface during the initial 4 cycles of testing resulted in an increase in transient activity and selectivity toward CO evolution.<sup>255</sup> The *pre-operando* Raman spectrum in Fig. 21e simultaneously revealed the generation of a reconstructive Co–O bond and the existence of an intermediate specie CO, elucidating the dynamic interplay between the catalytic site structure and the photocatalytic properties. The electronic structure of the reconstructed surface sites played a crucial role in modulating the CO<sub>2</sub> adsorption and CO desorption, ultimately reducing the energy barrier for the rate-determining step and enhancing CO<sub>2</sub> reduction activity and selectivity. For some 2D semiconductors, the ultrathin structure can also promote the associated dynamic transformations during the photocatalytic process. Taking ultrathin BiOCl (U-BOC) as an example, the exposed {001} facets were terminated with chlorine atoms rather than the conventionally expected oxygen atoms.<sup>246</sup> The ultrathin structure and low binding forces facilitated the outward migration of surface Cl atoms, leading to the formation of Bi, Cl, and O atoms with various valence states and corresponding chemical bond alterations in the lattice layers. During the photocatalytic CO<sub>2</sub>RR process, light irradiation initiated the retraction of surface Cl atoms, causing the elongation of Bi–Cl and Bi–O bonds in the lattice phase, along with the disappearance of multiple-valence Bi, Cl, and O atoms. With the assistance of the structural feature alteration described above, U-BOC exhibited outstanding photocatalytic CO<sub>2</sub> reduction performance, approximately 21.36 μmol g<sup>-1</sup> h<sup>-1</sup>, nearly 6.4 times higher than that of bulk BOC.

Constructing composite photocatalysts is an effective approach to enhance photocatalytic performance. Recently, a π–π stacking hybrid structure was formed between two highly conjugated materials: g-C<sub>3</sub>N<sub>4</sub> and the 2D-MOF moiety of the hybrid photocatalyst employed tetrakis(4-carboxylphenyl) porphyrin (TCPP) as the building block (Cu–CuTCPP), wherein g-C<sub>3</sub>N<sub>4</sub> served as the photosensitizer to provide electrons and 2D-

MOF Cu–CuTCPP acted as the interface to catalyze CO<sub>2</sub> reduction.<sup>245</sup> The node sites of the 2D-MOF were found to be crucial for C<sub>2</sub>H<sub>6</sub> production, and a dynamic transformation process during photocatalysis was observed: the initial paddle-wheel Cu<sub>2</sub>(COO)<sub>4</sub> node reconstructed into the partially reduced Cu<sup>1+δ</sup>(COO)<sub>3</sub>. This reconstruction enhanced the capture of *in situ* generated CO and the synergistic effects of the dual Cu sites, ultimately leading to efficient C–C coupling for C<sub>2</sub>H<sub>6</sub> production. Accordingly, this hybrid photocatalyst demonstrated remarkable efficiency in reducing CO<sub>2</sub> to C<sub>2</sub>H<sub>6</sub> with a selectivity of 44%, and the selectivity for total hydrocarbons (C<sub>2</sub>H<sub>6</sub> and CH<sub>4</sub>) reached 71%, making it one of the most promising photocatalytic systems reported to date. Based on this work, a Cu–ZnTCPP/g-C<sub>3</sub>N<sub>4</sub> hybrid photocatalyst was fabricated and it exhibited remarkable photocatalytic performance in converting CO<sub>2</sub> into CO and CH<sub>4</sub> under typical air conditions with 20% O<sub>2</sub>.<sup>219</sup> Interestingly, the presence of O<sub>2</sub> did not inhibit CO<sub>2</sub> reduction; instead, it led to a five-fold increase in the reaction rate compared to when O<sub>2</sub> was absent, as depicted in Fig. 21f. In the presence of O<sub>2</sub>, the Cu node in the 2D-MOF component experienced hydroxylation due to the hydroxyl radicals generated from O<sub>2</sub> reduction. Under these conditions, the hydroxylated Cu node retained its structure throughout aerobic CO<sub>2</sub> reduction but underwent structural changes and deactivation in the absence of O<sub>2</sub>. Further theoretical calculations verified a preference for CO<sub>2</sub> reduction over O<sub>2</sub> reduction on the hydroxylated Cu node (Fig. 22).

Comprehending and analyzing dynamic changes in heterojunction photocatalysts can be challenging, given the necessity to account for both the reaction environment and complex composition. For CuO<sub>x</sub>/TiO<sub>2</sub> photocatalysts, the different crystals of TiO<sub>2</sub> would influence the dynamic transformation of CuO<sub>x</sub> during the photocatalytic CO<sub>2</sub>RR.<sup>252</sup> Under Xe light irradiation, CuO species on TiO<sub>2</sub>-{001} were photoreduced to stable Cu<sub>2</sub>O species, promoting CO<sub>2</sub> reduction to CH<sub>4</sub>. However, CuO<sub>x</sub> species on TiO<sub>2</sub>-{100} and TiO<sub>2</sub>-{101} are photoreduced to Cu<sup>0</sup> species, hindering CH<sub>4</sub> production. In addition, each constituent in a heterojunction photocatalyst may undergo dynamic transformations. A typical example is the TiO<sub>2</sub>@BiOCl heterojunction photocatalyst.<sup>249</sup> Under light excitation, H-incorporated oxygen vacancies were formed on the defective TiO<sub>2</sub> nanotube matrix to extend the light adsorption range, and concurrent reduction of BiOCl lamina resulted in the formation of an atomic heterojunction composed of BiOCl–Bi<sup>0</sup>–TiO<sub>2</sub>. Bi<sup>0</sup> atoms directly transferred photogenerated electrons to Ti<sup>3+</sup> within H-incorporated oxygen vacancies, facilitating efficient electron gathering and thus enhancing photocatalytic efficiency. The presence of H-incorporated oxygen vacancies and Bi–Ti dual metal sites optimized the CO<sub>2</sub> reduction pathway, leading to high CH<sub>4</sub> selectivity. With the assistance of the dynamic reconstruction, a remarkable photocatalytic CO<sub>2</sub>RR performance was achieved by TiO<sub>2</sub>@BiOCl with the optimal CH<sub>4</sub> productivity rate of 168.5 μmol g<sup>-1</sup> h<sup>-1</sup> and CH<sub>4</sub> selectivity of 99.4%.

**4.2.3 Photocatalytic phenol degradation.** In addition to the water splitting and CO<sub>2</sub>RR mentioned above, dynamic



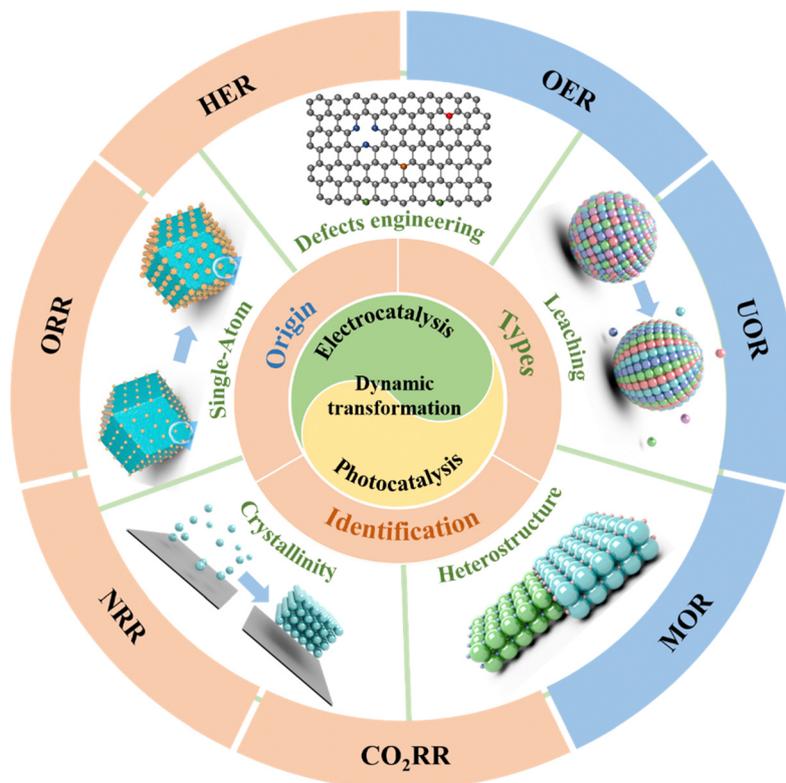


Fig. 22 Overview of dynamic transformation of active sites in the field of photo/electrocatalysis, covering the origin and types of dynamic transformation, identification methods, various triggering factors and applications in photo/electrocatalysis.

transformation has also been observed in many other photocatalytic reactions, such as pollutant degradation. Phenol and phenolic compounds are prevalent constituents in industrial wastewater effluents. The mixed Zn(Ga, Al)O oxides were produced *via* the thermal decomposition of ZnAl and ZnGaAl compounds featuring the LDH crystalline phase.<sup>252</sup> Throughout the photocatalytic phenol degradation process, these mixed oxide photocatalysts exhibited a partial restoration of their original layered structure, with a mixture of crystalline arrangements emerging after degradation. These mixed oxides outperformed the benchmark Degussa P25 TiO<sub>2</sub>, degrading nearly 60% of the initial 80 ppm phenol within 6 hours, while P25 TiO<sub>2</sub> achieved only about 25% degradation. In another research, the chemical state transformation of Ce (from Ce<sup>4+</sup> to Ce<sup>3+</sup>) in Ce-doped ZnAl LDH materials during photocatalytic phenol degradation promoted the separation of photogenerated electrons and holes.<sup>253</sup> This was attributed to the presence of Ce in different oxidation states (Ce<sup>4+</sup> and Ce<sup>3+</sup>), which acted as electron and hole receptors, preventing their recombination.

In most cases, the comprehension of photocatalysis is attained by evaluating photocatalysts before and after photocatalytic reactions, thereby establishing an empirical foundation for deducing the photocatalytic processes. However, the photocatalytic mechanism remains elusive due to the involvement of high-energy electrons and holes, which act as “reactants” in photocatalytic reactions. These photogenerated charges not only participate in surface catalytic reactions at

the outer surface but also potentially trigger reconstruction or self-activation of photocatalysts before reaching the outer surface. In comparison to electrocatalysts, dynamic transformation studies on photocatalysts are relatively limited and primarily concentrate on photocatalytic water splitting and the CO<sub>2</sub>RR. Future research endeavors should prioritize investigating dynamic changes in photocatalysts for a broad range of reactions. This approach will contribute to a comprehensive understanding of photocatalytic mechanisms.

## 5. Conclusion and perspectives

Over recent decades, there has been significant progress in the development of catalyst dynamic transformation, encompassing the exploration of the origins, underlying principles, identification methods, and targeted tailoring of the dynamic transformations. These advancements offer substantial potential for engineering the dynamic transformation to design efficient and durable photo/electrocatalysts. This comprehensive overview systematically explored such issues and identified several approaches and important triggering factors for the dynamic transformation and its applications in photo/electrocatalysis. It should be noted that reconstruction of catalysts is induced by single atom dopants, defects (vacancies, dopants, grain boundaries, *etc.*), heterostructures, leaching and crystallinity, which correspond to scales ranging from the atomistic



defects to the mesoscopic scale crystalline structures of the materials. Despite the advancements, constructing photo/electrocatalysts with precisely controlled reconstructions for diverse photo/electrocatalytic reactions remains challenging. Enhancing the activity, stability and selectivity of these reconstructed photo/electrocatalysts for specific reactions poses ongoing challenges. Several critical questions remain:

(1) Difficulty in *in situ* characterization: the dynamic environment of electrochemical reactions during catalyst restructuring poses significant challenges in accurately characterizing these changes. In particular, the study of active sites has focused on individual atoms, coordination structures, and specific crystal facets, so the corresponding characterization methods need to be further developed, which is conducive to the systematic description and understanding of the reconstruction mechanism. For example, researchers initially recognized the reconstruction process through changes in cyclic voltammetry curves. Subsequently, advancements in characterization techniques have evolved considerably, such as *in situ* TEM for observing surface morphological changes and *in situ* XAS for obtaining coordination environments and electronic structure information. The evolution of characterization techniques facilitates the correlation of dynamic changes in active sites with catalytic activity, product selectivity, and stability. This capability effectively guides the regulation of photo/electrocatalysts reconstruction.

(2) Catalyst restructuring involves complex chemical reaction pathways and intermediates. Identifying the true active sites and understanding the reaction mechanisms of dynamic processes are daunting and critical challenges in the field of catalysis. For instance, owing to the iron's inherent propensity for active reconstruction, the extensive research endeavors have been focused on unraveling the dynamic restructuring phenomena associated with iron, particularly within the realm of NiFe (hydroxide) oxides. Despite these efforts, a consensus on the dynamic transformation pathways of iron remains elusive, particularly concerning alterations in its oxidation states and coordination environments. Similarly, in the realm of photo/electrocatalysts for the CO<sub>2</sub>RR, predominantly Cu-based, the significance of discerning active sites cannot be overstated. These sites must efficiently adsorb CO<sub>2</sub> molecules and facilitate the coupling of carbon atoms to yield multicarbon products, where material selectivity profoundly influences catalyst restructuring. Factors such as varying oxidation states of copper sites, distinct crystal facets, and specialized structures resulting from catalyst reconstruction, significantly impact the selectivity of reduction products. Hence, the imperative lies in recognizing the authentic catalytic sites and delving into the dynamics of the process. Such endeavors not only contribute to bolstering our comprehension of catalytic mechanisms but also pave the way for the development of more efficient and selective photo/electrocatalysts.

(3) Rationally controlling the degree of reconstruction in catalytic systems is of paramount importance for achieving a delicate balance between stability and activity. This equilibrium is crucial for optimizing catalytic performance and ensuring

long-term efficacy. By strategically modulating the reconstruction process, one can tailor the surface structure and composition of photo/electrocatalysts to enhance their stability while preserving their catalytic activity. For example, noble metal oxide catalysts exhibit high activity in the OER due to their tendency to follow the LOM mechanism, which typically involves lower thermodynamic barriers. However, this heightened activity often leads to irreversible site dissolution, resulting in decreased stability of the catalyst. To address this issue, controlling the degree of self-reconstruction through modification of noble metal catalysts is crucial. Precise modulation of the reconstruction process allows for optimization of the surface structure and composition of the catalysts, thereby enhancing both activity and stability. Modification strategies may include introducing auxiliary species with higher stability, tuning the lattice structure or surface adsorption properties of the photo/electrocatalysts, and designing surface coatings resistant to dissolution.

(4) While there has been considerable discourse on the reconstruction mechanism in the electrocatalytic process, there is a notable scarcity of reports investigating the reconstruction mechanism under photocatalytic conditions. Moreover, the oxidation half-reaction is frequently overlooked in many instances, despite the equal significance of both the reduction and oxidation half-reactions. The rational and strategic utilization of dynamic transformation in the reduction reaction, particularly in photocatalysis, should be a focal point in future research efforts. Investigating the structure–performance relationship under operational conditions not only aids the identification of active sites/species but also enables the development of highly active and selective photo/electrocatalysts, leading to enhanced production rates and yields. For practical applications, the search for cost-effective, abundantly available, and atom-efficient catalytic materials is imperative, with a particular emphasis on single-atom/single-site catalysts, especially those based on non-precious metals such as Fe, Co, Ni, and Mn, to be guided by the further established structure–activity correlations.

(5) When considering the integration of catalysts into electrolyzers, exploring catalyst reconstruction involves the synergy of multiple factors. For instance, under alkaline industrial electrolysis conditions, where the electrolyte concentration typically ranges from 20 to 30 wt% KOH at temperatures of 50–80 °C, the catalyst's reconstruction characteristics become pivotal. Additionally, the evolution of catalyst-ionomer mixtures within MEA under electrolysis conditions represents another critical yet underreported issue. For photocatalysis, the development of a mature device for the large-scale application of photocatalytic reactions remains an ongoing challenge. In recent years, significant research has been dedicated to investigating various photocatalytic devices, each with its own unique reaction environment. For example, under controlled laboratory conditions, photocatalytic water splitting is frequently conducted using suspended powder photocatalysts directly in aqueous solutions.<sup>236–238</sup> However, when transitioning to large-scale applications, it becomes imperative to deposit



the catalyst onto a specific substrate. Such different reaction environments may cause differences in the dynamic transformation of photocatalysts during the reaction process.

(6) Integrating high-throughput experimentation and machine learning for studying catalyst reconstruction: high-throughput experimentation (HTE) enables the rapid screening and optimization of a large number of catalyst samples particularly in studying reconstruction behaviors of different elements and structures under various conditions. Machine learning algorithms can analyze vast datasets generated from HTE to uncover patterns and relationships that might be missed by traditional methods. By training AI models on experimental data, researchers can develop predictive models that provide insights into catalyst reconstruction behaviors and guide future experiments.

## Author contributions

H. Zhang conceived the idea. H. Zhang, L. Chen and F. Dong co-wrote the paper. H. Zhang, Z. Lu, E. Lv, X. Dong, H. Li, Z. Yuan, X. Peng, S. Yang, J. Qiu, Z. Guo and Z. Wen discussed the results and revised the manuscript.

## Data availability

The authors declare that the data supporting the findings of this study are available in the article.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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## References

- L. Liu, G. He, M. Wu, G. Liu, H. Zhang, Y. Chen, J. Shen and S. Li, *Nat. Energy*, 2023, **8**, 870–880.
- M. Rastgar, K. Moradi, C. Burroughs, A. Hemmati, E. Hoek and M. Sadrzadeh, *Chem. Rev.*, 2023, **123**, 10156–10205.
- M. A. Green, *Nat. Energy*, 2023, **8**, 783–784.
- Q. Wang, C. Pornrungrroj, S. Linley and E. Reisner, *Nat. Energy*, 2022, **7**, 13–24.
- J. Ran, A. Talebian-Kiakalaieh, S. Zhang, E. M. Hashem, M. Guo and S.-Z. Qiao, *Chem. Sci.*, 2024, **15**, 1611–1637.
- V. R. Stamenkovic, D. Strmcnik, P. P. Lopes and N. M. Markovic, *Nat. Mater.*, 2017, **16**, 57–69.
- A. Buttler and H. Spliethoff, *Renewable Sustainable Energy Rev.*, 2018, **82**, 2440–2454.
- W. Cheng, X. Zhao, H. Su, F. Tang, W. Che, H. Zhang and Q. Liu, *Nat. Energy*, 2019, **4**, 115–122.
- H. Sun, Z. Yan, F. Liu, W. Xu, F. Cheng and J. Chen, *Adv. Mater.*, 2020, **32**, 1806326.
- W. Fang, W. Guo, R. Lu, Y. Yan, X. Liu, D. Wu, F. M. Li, Y. Zhou, C. He and C. Xia, *Nature*, 2024, **626**, 86–91.
- M. Zheng, J. Zhang, P. Wang, H. Jin, Y. Zheng and S. Z. Qiao, *Adv. Mater.*, 2024, **36**, 2307913.
- H. Jiang, Q. He, Y. Zhang and L. Song, *Acc. Chem. Res.*, 2018, **51**, 2968–2977.
- Y. Li, X. Du, J. Huang, C. Wu, Y. Sun, G. Zou, C. Yang and J. Xiong, *Small*, 2019, **15**, 1901980.
- H. S. Taylor, *Proceedings of the Royal Society of London. Series A, Containing Papers of a Mathematical and Physical Character*, 1925, **108**, 105–111.
- C. Pei, S. Chen, D. Fu, Z.-J. Zhao and J. Gong, *Chem. Rev.*, 2024, **124**, 2955–3012.
- W. Chen, Y. Che, J. Xia, L. Zheng, H. Lv, J. Zhang, H.-W. Liang, X. Meng, D. Ma, W. Song, X. Wu and C. Cao, *J. Am. Chem. Soc.*, 2024, **146**, 11542–11552.
- M. N. Jackson, C. J. Kaminsky, S. Oh, J. F. Melville and Y. Surendranath, *J. Am. Chem. Soc.*, 2019, **141**, 14160–14167.
- J.-W. Zhao, H.-Y. Wang, L. Feng, J.-Z. Zhu, J.-X. Liu and W.-X. Li, *Chem. Rev.*, 2024, **124**, 164–209.
- D. Liu, Q. He, S. Ding and L. Song, *Adv. Energy Mater.*, 2020, **10**, 2001482.
- R. Chen, Z. Ren, Y. Liang, G. Zhang, T. Dittrich, R. Liu, Y. Liu, Y. Zhao, S. Pang, H. An, C. Ni, P. Zhou, K. Han, F. Fan and C. Li, *Nature*, 2022, **610**, 296–301.
- T. Tang, X. Bai, Z. Wang and J. Guan, *Chem. Sci.*, 2024, **15**, 5082–5112.
- E. Pastor, M. Sachs, S. Selim, J. R. Durrant, A. A. Bakulin and A. Walsh, *Nat. Rev. Mater.*, 2022, **7**, 503–521.
- J. Ren, H. Chi, L. Tan, Y.-K. Peng, G. Li, M. Meng-Jung Li, Y. Zhao and X. Duan, *J. Mater. Chem. A*, 2023, **11**, 2528–2543.
- J. Li, Y. Zhang, K. Kuruvinashetti and N. Kornienko, *Nat. Rev. Chem.*, 2022, **6**, 303–319.
- H.-Q. Chen, L. Zou, D.-Y. Wei, L.-L. Zheng, Y.-F. Wu, H. Zhang and J.-F. Li, *Chin. J. Catal.*, 2022, **43**, 33–46.



- 26 S. Zhao, C. Tan, C.-T. He, P. An, F. Xie, S. Jiang, Y. Zhu, K.-H. Wu, B. Zhang and H. Li, *Nat. Energy*, 2020, **5**, 881–890.
- 27 B. Zhang, L. Wang, Z. Cao, S. M. Kozlov, F. P. García de Arquer, C. T. Dinh, J. Li, Z. Wang, X. Zheng and L. Zhang, *Nat. Catal.*, 2020, **3**, 985–992.
- 28 Q. Liu, Q. Jiang, L. Li and W. Yang, *J. Am. Chem. Soc.*, 2024, **146**, 4242–4251.
- 29 K. Seki, T. Higashi, Y. Kawase, K. Takanabe and K. Domen, *J. Phys. Chem. Lett.*, 2022, **13**, 10356–10363.
- 30 F.-M. Li, L. Huang, S. Zaman, W. Guo, H. Liu, X. Guo and B. Y. Xia, *Adv. Mater.*, 2022, **34**, 2200840.
- 31 L. Wang, Y. Hao, L. Deng, F. Hu, S. Zhao, L. Li and S. Peng, *Nat. Commun.*, 2022, **13**, 5785.
- 32 Y. Xie, Y. Sun, H. Tao, X. Wang, J. Wu, K. Ma, L. Wang, Z. Kang and Y. Zhang, *Adv. Funct. Mater.*, 2022, **32**, 2111777.
- 33 Y. Kuang, M. J. Kenney, Y. Meng, W.-H. Hung, Y. Liu, J. E. Huang, R. Prasanna, P. Li, Y. Li and L. Wang, *Proc. Natl. Acad. Sci. U. S. A.*, 2019, **116**, 6624–6629.
- 34 X. Liu, J. Meng, J. Zhu, M. Huang, B. Wen, R. Guo and L. Mai, *Adv. Mater.*, 2021, **33**, 2007344.
- 35 N. Ran, E. Song, Y. Wang, Y. Zhou and J. Liu, *Energy Environ. Sci.*, 2022, **15**, 2071–2083.
- 36 K. Jeenuang, P. Pornaroontham, M. F. Qureshi, P. Linga and P. Rangsunvigit, *Chem. Eng. J.*, 2024, **479**, 147691.
- 37 L. Liu and A. Corma, *Nat. Rev. Chem.*, 2021, **5**, 256–276.
- 38 J. Li, R. Chen, J. Wang, K. Wang, Y. Zhou, M. Xing and F. Dong, *Angew. Chem., Int. Ed.*, 2024, **63**, e202317575.
- 39 A. W. Heard, J. M. Suárez and S. M. Goldup, *Nat. Rev. Chem.*, 2022, **6**, 182–196.
- 40 D. A. Kuznetsov, B. Han, Y. Yu, R. R. Rao, J. Hwang, Y. Román-Leshkov and Y. Shao-Horn, *Joule*, 2018, **2**, 225–244.
- 41 G. Chen, C. Wei, Y. Zhu and H. Huang, *EcoMat*, 2023, **5**, e12294.
- 42 Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov and T. F. Jaramillo, *Science*, 2017, **355**, eaad4998.
- 43 Y. A. Alsunni and C. B. Musgrave, *Appl. Surf. Sci.*, 2023, **610**, 155147.
- 44 X. Luo, X. Tan, P. Ji, L. Chen, J. Yu and S. Mu, *EnergyChem*, 2023, **5**, 100091.
- 45 B. Yan, D. Krishnamurthy, C. H. Hendon, S. Deshpande, Y. Surendranath and V. Viswanathan, *Joule*, 2017, **1**, 600–612.
- 46 D. Gao, I. T. McCrum, S. Deo, Y.-W. Choi, F. Scholten, W. Wan, J. G. Chen, M. J. Janik and B. Roldan Cuenya, *ACS Catal.*, 2018, **8**, 10012–10020.
- 47 B.-H. Lee, S. Park, M. Kim, A. K. Sinha, S. C. Lee, E. Jung, W. J. Chang, K.-S. Lee, J. H. Kim and S.-P. Cho, *Nat. Mater.*, 2019, **18**, 620–626.
- 48 X. Li, C.-S. Cao, S.-F. Hung, Y.-R. Lu, W. Cai, A. I. Rykov, S. Miao, S. Xi, H. Yang and Z. Hu, *Chemistry*, 2020, **6**, 3440–3454.
- 49 J. Vavra, T. H. Shen, D. Stoian, V. Tileli and R. Buonsanti, *Angew. Chem., Int. Ed.*, 2021, **133**, 1367–1374.
- 50 X. Zou, Y. Wu, Y. Liu, D. Liu, W. Li, L. Gu, H. Liu, P. Wang, L. Sun and Y. Zhang, *Chemistry*, 2018, **4**, 1139–1152.
- 51 K. Fan, H. Zou, Y. Lu, H. Chen, F. Li, J. Liu, L. Sun, L. Tong, M. F. Toney and M. Sui, *ACS Nano*, 2018, **12**, 12369–12379.
- 52 R. Chen, S. Pang, H. An, J. Zhu, S. Ye, Y. Gao, F. Fan and C. Li, *Nat. Energy*, 2018, **3**, 655–663.
- 53 X. Song, Y. Wu, X. Zhang, X. Li, Z. Zhu, C. Ma, Y. Yan, P. Huo and G. Yang, *Chem. Eng. J.*, 2021, **408**, 127292.
- 54 L. Zhang, B. K. Miller and P. A. Crozier, *Nano Lett.*, 2013, **13**, 679–684.
- 55 W. Bi, X. Li, L. Zhang, T. Jin, L. Zhang, Q. Zhang, Y. Luo, C. Wu and Y. Xie, *Nat. Commun.*, 2015, **6**, 8647.
- 56 J. Hu, J. Xie, W. Jia, S. Zhang, S. Wang, K. Wang and Y. Cao, *Appl. Catal., B*, 2020, **268**, 118753.
- 57 A. Dutta, A. Kuzume, M. Rahaman, S. Vesztergom and P. Broekmann, *ACS Catal.*, 2015, **5**, 7498–7502.
- 58 S. Karasawa, K. P. Sharma, D. Yamamoto, T. Saida, S. Naritsuka and T. Maruyama, *Chem. Phys. Lett.*, 2022, **804**, 139889.
- 59 X. She, L. Zhai, Y. Wang, P. Xiong, M. M.-J. Li, T.-S. Wu, M. C. Wong, X. Guo, Z. Xu and H. Li, *Nat. Energy*, 2024, **1–11**.
- 60 W. A. Waters, *J. Am. Oil Chem. Soc.*, 1971, **48**, 427–433.
- 61 Y. Pan, Y. Li, C.-L. Dong, Y.-C. Huang, J. Wu, J. Shi, Y. Lu, M. Yang, S. Wang and Y. Zou, *Chemistry*, 2023, **9**, 963–977.
- 62 J.-J. Velasco-Velez, R. V. Mom, L.-E. Sandoval-Diaz, L. J. Falling, C.-H. Chuang, D. Gao, T. E. Jones, Q. Zhu, R. Arrigo and B. Roldan Cuenya, *ACS Energy Lett.*, 2020, **5**, 2106–2111.
- 63 Y. H. Li, C. Li and H. G. Yang, *J. Mater. Chem. A*, 2017, **5**, 20631–20634.
- 64 L. Quan, H. Jiang, G. Mei, Y. Sun and B. You, *Chem. Rev.*, 2024, **124**, 3694–3812.
- 65 S. Wang, W. Huo, F. Fang, Z. Xie, J. K. Shang and J. Jiang, *Chem. Eng. J.*, 2022, **429**, 132410.
- 66 J. Zhou, L. Zhang, Y.-C. Huang, C.-L. Dong, H.-J. Lin, C.-T. Chen, L. Tjeng and Z. Hu, *Nat. Commun.*, 2020, **11**, 1984.
- 67 K. Han, T. Kreuger, B. Mei and G. Mul, *ACS Catal.*, 2017, **7**, 1610–1614.
- 68 L. Chen, J.-T. Ren, Y.-S. Wang, W.-W. Tian, L.-J. Gao and Z.-Y. Yuan, *ACS Sustainable Chem. Eng.*, 2019, **7**, 13559–13568.
- 69 L. Chen, J.-T. Ren and Z.-Y. Yuan, *ACS Sustainable Chem. Eng.*, 2021, **9**, 1856–1866.
- 70 P. Strasser and S. Kühn, *Nano Energy*, 2016, **29**, 166–177.
- 71 H. L. Xin, J. A. Mundy, Z. Liu, R. Cabezas, R. Hovden, L. F. Kourkoutis, J. Zhang, N. P. Subramanian, R. Makharia and F. T. Wagner, *Nano Lett.*, 2012, **12**, 490–497.
- 72 L. Zhu, B. Wei, Y. Zhang, Z. Lü, Z. Wang, X. Huang, Z. Cao, W. Jiang and Y. Li, *Electrochim. Acta*, 2015, **160**, 89–93.
- 73 Y. Wang, Y. Yang, S. Jia, X. Wang, K. Lyu, Y. Peng, H. Zheng, X. Wei, H. Ren and L. Xiao, *Nat. Commun.*, 2019, **10**, 1506.
- 74 A. Dhakshinamoorthy, A. M. Asiri and H. Garcia, *Angew. Chem., Int. Ed.*, 2016, **55**, 5414–5445.
- 75 W. Zheng, M. Liu and L. Y. S. Lee, *ACS Catal.*, 2019, **10**, 81–92.
- 76 A. Corma, P. Concepción, M. Boronat, M. J. Sabater, J. Navas, M. J. Yacamán, E. Larios, A. Posadas,



- M. A. López-Quintela and D. Buceta, *Nat. Chem.*, 2013, **5**, 775–781.
- 77 D. Y. Chung, P. P. Lopes, P. Farinazzo Bergamo Dias Martins, H. He, T. Kawaguchi, P. Zapol, H. You, D. Tripkovic, D. Strmcnik and Y. Zhu, *Nat. Energy*, 2020, **5**, 222–230.
- 78 M. Schwarz, F. Faisal, S. Mohr, C. Hohner, K. Werner, T. Xu, T. S. Skála, N. Tsud, K. C. Prince and V. Matolín, *J. Phys. Chem. Lett.*, 2018, **9**, 2763–2769.
- 79 E. Fabbri, M. Nachtegaal, T. Binninger, X. Cheng, B.-J. Kim, J. Durst, F. Bozza, T. Graule, R. Schäublin and L. Wiles, *Nat. Mater.*, 2017, **16**, 925–931.
- 80 C. Sun, H. Zhang, H. Liu, X. Zheng, W. Zou, L. Dong and L. Qi, *Appl. Catal., B*, 2018, **235**, 66–74.
- 81 T. K. Townsend, N. D. Browning and F. E. Osterloh, *Energy Environ. Sci.*, 2012, **5**, 9543–9550.
- 82 Y. Yu, H. L. Xin, R. Hovden, D. Wang, E. D. Rus, J. A. Mundy, D. A. Muller and H. D. Abruña, *Nano Lett.*, 2012, **12**, 4417–4423.
- 83 Z. Kou, Y. Yu, X. Liu, X. Gao, L. Zheng, H. Zou, Y. Pang, Z. Wang, Z. Pan and J. He, *ACS Catal.*, 2020, **10**, 4411–4419.
- 84 X. Fang, Q. Shang, Y. Wang, L. Jiao, T. Yao, Y. Li, Q. Zhang, Y. Luo and H. L. Jiang, *Adv. Mater.*, 2018, **30**, 1705112.
- 85 X. Li, W. Bi, L. Zhang, S. Tao, W. Chu, Q. Zhang, Y. Luo, C. Wu and Y. Xie, *Adv. Mater.*, 2016, **28**, 2427–2431.
- 86 C. Dessal, L. Martínez, C. Maheu, T. Len, F. Morfin, J. L. Rousset, E. Puzenat, P. Afanasiev, M. Aouine and L. Soler, *J. Catal.*, 2019, **375**, 155–163.
- 87 L. Liu, D. M. Meira, R. Arenal, P. Concepcion, A. V. Puga and A. Corma, *ACS Catal.*, 2019, **9**, 10626–10639.
- 88 L. Liao, Q. Zhang, Z. Su, Z. Zhao, Y. Wang, Y. Li, X. Lu, D. Wei, G. Feng and Q. Yu, *Nat. Nanotechnol.*, 2014, **9**, 69–73.
- 89 T. K. Todorova, M. W. Schreiber and M. Fontecave, *ACS Catal.*, 2019, **10**, 1754–1768.
- 90 P. Grosse, D. Gao, F. Scholten, I. Sinev, H. Mistry and B. Roldan Cuenya, *Angew. Chem., Int. Ed.*, 2018, **130**, 6300–6305.
- 91 F. Guo, Y. Wu, H. Chen, Y. Liu, L. Yang, X. Ai and X. Zou, *Energy Environ. Sci.*, 2019, **12**, 684–692.
- 92 J. Huang, Y. Li, Y. Zhang, G. Rao, C. Wu, Y. Hu, X. Wang, R. Lu, Y. Li and J. Xiong, *Angew. Chem., Int. Ed.*, 2019, **131**, 17619–17625.
- 93 Y. Yang, Y. Xiong, R. Zeng, X. Lu, M. Krumov, X. Huang, W. Xu, H. Wang, F. J. DiSalvo and J. D. Brock, *ACS Catal.*, 2021, **11**, 1136–1178.
- 94 J. Timoshenko and B. Roldan Cuenya, *Chem. Rev.*, 2020, **121**, 882–961.
- 95 G. Xing, M. Tong, P. Yu, L. Wang, G. Zhang, C. Tian and H. Fu, *Angew. Chem., Int. Ed.*, 2022, **61**, e202211098.
- 96 Z. Yan, H. Sun, X. Chen, H. Liu, Y. Zhao, H. Li, W. Xie, F. Cheng and J. Chen, *Nat. Commun.*, 2018, **9**, 2373.
- 97 X. Han, C. Yu, Y. Niu, Z. Wang, Y. Kang, Y. Ren, H. Wang, H. S. Park and J. Qiu, *Small Methods*, 2020, **4**, 2000546.
- 98 B. Wang, K. Zhao, Z. Yu, C. Sun, Z. Wang, N. Feng, L. Mai, Y. Wang and Y. Xia, *Energy Environ. Sci.*, 2020, **13**, 2200–2208.
- 99 L. Gao, X. Cui, Z. Wang, C. Sewell, Z. Li, S. Liang and M. Zhang, *Proceedings of the National Academy of Sciences*, 2021, **118**, e2023421118.
- 100 X. Wang, K. Klingan, M. Klingenhof, T. Möller, J. Ferreira de Araújo, I. Martens, A. Bagger, S. Jiang, J. Rossmeisl and H. Dau, *Nat. Commun.*, 2021, **12**, 794.
- 101 T.-H. Shen, L. Spillane, J. Vavra, T. H. M. Pham, J. Peng, Y. Shao-Horn and V. Tileli, *J. Am. Chem. Soc.*, 2020, **142**, 15876–15883.
- 102 P. Grosse, D. Gao, F. Scholten, I. Sinev, H. Mistry and B. R. Cuenya, *Angew. Chem., Int. Ed.*, 2018, **130**, 6300–6305.
- 103 Y. Gogotsi, *Nat. Mater.*, 2015, **14**, 1079–1080.
- 104 C. Ling, L. Shi, Y. Ouyang, Q. Chen and J. Wang, *Adv. Sci.*, 2016, **3**, 1600180.
- 105 C. Xu, L. Wang, Z. Liu, L. Chen, J. Guo, N. Kang, X.-L. Ma, H.-M. Cheng and W. Ren, *Nat. Mater.*, 2015, **14**, 1135–1141.
- 106 L. Liu and A. Corma, *Trends Chem.*, 2020, **2**, 383–400.
- 107 J. Li, Q. Guan, H. Wu, W. Liu, Y. Lin, Z. Sun, X. Ye, X. Zheng, H. Pan and J. Zhu, *J. Am. Chem. Soc.*, 2019, **141**, 14515–14519.
- 108 M. D. Marcinkowski, S. F. Yuk, N. Doudin, R. S. Smith, M.-T. Nguyen, B. D. Kay, V.-A. Glezakou, R. Rousseau and Z. Dohnálek, *ACS Catal.*, 2019, **9**, 10977–10982.
- 109 Z. Huang, T. Ban, Y. Zhang, L. Wang, S. Guo, C.-R. Chang and G. Jing, *Appl. Catal., B*, 2021, **283**, 119625.
- 110 X. Liang, N. Fu, S. Yao, Z. Li and Y. Li, *J. Am. Chem. Soc.*, 2022, **144**, 18155–18174.
- 111 X. Wei, Y. Liu, X. Zhu, S. Bo, L. Xiao, C. Chen, T. T. T. Nga, Y. He, M. Qiu and C. Xie, *Adv. Mater.*, 2023, **35**, 2300020.
- 112 D. Karapinar, N. T. Huan, N. Ranjbar Sahraie, J. Li, D. Wakerley, N. Touati, S. Zanna, D. Taverna, L. H. Galvão Tizei and A. Zitolo, *Angew. Chem., Int. Ed.*, 2019, **58**, 15098–15103.
- 113 J. Yang, H. Qi, A. Li, X. Liu, X. Yang, S. Zhang, Q. Zhao, Q. Jiang, Y. Su and L. Zhang, *J. Am. Chem. Soc.*, 2022, **144**, 12062–12071.
- 114 M. Tong, F. Sun, G. Xing, C. Tian, L. Wang and H. Fu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202314933.
- 115 H. Sun, C.-W. Tung, Y. Qiu, W. Zhang, Q. Wang, Z. Li, J. Tang, H.-C. Chen, C. Wang and H. M. Chen, *J. Am. Chem. Soc.*, 2021, **144**, 1174–1186.
- 116 Y. Zhou, W. Zhang, J. Hu, D. Li, X. Yin and Q. Gao, *ACS Sustainable Chem. Eng.*, 2021, **9**, 7390–7399.
- 117 J. Meng, Z. Lan, I. E. Castelli and K. Zheng, *J. Phys. Chem. C*, 2021, **125**, 8456–8460.
- 118 D. Zhou, F. Li, Y. Zhao, L. Wang, H. Zou, Y. Shan, J. Fu, Y. Ding, L. Duan and M. Liu, *ACS Catal.*, 2023, **13**, 4398–4408.
- 119 L. Peng, N. Yang, Y. Yang, Q. Wang, X. Xie, D. Sun-Waterhouse, L. Shang, T. Zhang and G. I. Waterhouse, *Angew. Chem., Int. Ed.*, 2021, **60**, 24612–24619.
- 120 Q. He, Y. Wan, H. Jiang, Z. Pan, C. Wu, M. Wang, X. Wu, B. Ye, P. M. Ajayan and L. Song, *ACS Energy Lett.*, 2018, **3**, 1373–1380.
- 121 P. Chen, T. Zhou, S. Wang, N. Zhang, Y. Tong, H. Ju, W. Chu, C. Wu and Y. Xie, *Angew. Chem., Int. Ed.*, 2018, **57**, 15471–15475.



- 122 W. Peng, J. Li, K. Shen, L. Zheng, H. Tang, Y. Gong, J. Zhou, N. Chen, S. Zhao and M. Chen, *J. Mater. Chem. A*, 2020, **8**, 23580–23589.
- 123 J. Bai, J. Mei, T. Liao, Q. Sun, Z. G. Chen and Z. Sun, *Adv. Energy Mater.*, 2022, **12**, 2103247.
- 124 C. Hu, K. Yue, J. Han, X. Liu, L. Liu, Q. Liu, Q. Kong, C.-W. Pao, Z. Hu and K. Suenaga, *Sci. Adv.*, 2023, **9**, eadf9144.
- 125 S. Tao, G. Zhang, B. Qian, J. Yang, S. Chu, C. Sun, D. Wu, W. Chu and L. Song, *Appl. Catal., B*, 2023, **330**, 122600.
- 126 Y. Wei, Y. Zheng, Y. Hu, B. Huang, M. Sun, P. Da, P. Xi and C.-H. Yan, *ACS Appl. Mater. Interfaces*, 2022, **14**, 25638–25647.
- 127 H. Liao, C. Wei, J. Wang, A. Fisher, T. Sritharan, Z. Feng and Z. J. Xu, *Adv. Energy Mater.*, 2017, **7**, 1701129.
- 128 Z. Xiao, M. Yang, C. Liu, B. Wang, S. Zhang, J. Liu, Z. Xu, R. Gao, J.-J. Zou and A. Tang, *Nano Energy*, 2022, **98**, 107233.
- 129 Y. Tang, C. Wu, Q. Zhang, H. Zhong, A. Zou, J. Li, Y. Ma, H. An, Z. Yu and S. Xi, *Angew. Chem., Int. Ed.*, 2023, **135**, e202309107.
- 130 S. Zhang, S. Gu, Y. Wang, C. Liang, Y. Yu, L. Han, S. Zheng, N. Zhang, X. Liu and J. Zhou, *ACS Catal.*, 2019, **9**, 7389–7397.
- 131 H. Jiang, Q. He, X. Li, X. Su, Y. Zhang, S. Chen, S. Zhang, G. Zhang, J. Jiang and Y. Luo, *Adv. Mater.*, 2019, **31**, 1805127.
- 132 A. L. Maulana, P.-C. Chen, Z. Shi, Y. Yang, C. Lizandara-Pueyo, F. Seeler, H. D. Abruña, D. Muller, K. Schierle-Arndt and P. Yang, *Nano Lett.*, 2023, **23**, 6637–6644.
- 133 M. Ning, F. Zhang, L. Wu, X. Xing, D. Wang, S. Song, Q. Zhou, L. Yu, J. Bao and S. Chen, *Energy Environ. Sci.*, 2022, **15**, 3945–3957.
- 134 R. Yu, C. Wang, D. Liu, X. Wang, J. Yin and Y. Du, *Inorg. Chem.*, 2022, **62**, 609–617.
- 135 L. Qiu, Q. Wang, P. Yan and X.-Y. Yu, *J. Mater. Chem. A*, 2022, **10**, 21251–21259.
- 136 P. Yan, Q. Liu, H. Zhang, L. Qiu, H. B. Wu and X.-Y. Yu, *J. Mater. Chem. A*, 2021, **9**, 15586–15594.
- 137 C. Kuai, C. Xi, A. Hu, Y. Zhang, Z. Xu, D. Nordlund, C.-J. Sun, C. A. Cadigan, R. M. Richards and L. Li, *J. Am. Chem. Soc.*, 2021, **143**, 18519–18526.
- 138 F. Yang, M. Lopez Luna, F. T. Haase, D. Escalera-López, A. Yoon, M. Rüscher, C. Rettenmaier, H. S. Jeon, E. Ortega and J. Timoshenko, *J. Am. Chem. Soc.*, 2023, **145**, 21465–21474.
- 139 D. Yao, C. Tang, A. Vasileff, X. Zhi, Y. Jiao and S. Z. Qiao, *Angew. Chem., Int. Ed.*, 2021, **133**, 18326–18332.
- 140 W. Cai, R. Chen, H. Yang, H. B. Tao, H.-Y. Wang, J. Gao, W. Liu, S. Liu, S.-F. Hung and B. Liu, *Nano Lett.*, 2020, **20**, 4278–4285.
- 141 Y. Duan, Z. Y. Yu, S. J. Hu, X. S. Zheng, C. T. Zhang, H. H. Ding, B. C. Hu, Q. Q. Fu, Z. L. Yu and X. Zheng, *Angew. Chem., Int. Ed.*, 2019, **58**, 15772–15777.
- 142 G. Chen, Z. Hu, Y. Zhu, B. Gu, Y. Zhong, H. J. Lin, C. T. Chen, W. Zhou and Z. Shao, *Adv. Mater.*, 2018, **30**, 1804333.
- 143 Z. Ma, C. Chen, X. Cui, L. Zeng, L. Wang, W. Jiang and J. Shi, *ACS Appl. Mater. Interfaces*, 2021, **13**, 44224–44233.
- 144 Y. Wang, Y. Zhao, L. Liu, W. Qin, S. Liu, J. Tu, Y. Liu, Y. Qin, J. Liu and H. Wu, *J. Am. Chem. Soc.*, 2023, **145**, 20261–20272.
- 145 Z. P. Wu, X. F. Lu, S. Q. Zang and X. W. Lou, *Adv. Funct. Mater.*, 2020, **30**, 1910274.
- 146 F.-S. Zhang, J.-W. Wang, J. Luo, R.-R. Liu, Z.-M. Zhang, C.-T. He and T.-B. Lu, *Chem. Sci.*, 2018, **9**, 1375–1384.
- 147 H. Li, S. Wang, H. Sawada, G. G. Han, T. Samuels, C. S. Allen, A. I. Kirkland, J. C. Grossman and J. H. Warner, *ACS Nano*, 2017, **11**, 3392–3403.
- 148 E. Mosconi, D. Meggiolaro, H. J. Snaith, S. D. Stranks and F. De Angelis, *Energy Environ. Sci.*, 2016, **9**, 3180–3187.
- 149 R. A. De Souza, V. Metlenko, D. Park and T. E. Weirich, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 174109.
- 150 Y. Li, P. Hasin and Y. Wu, *Adv. Mater.*, 2010, **17**, 1926–1929.
- 151 S. Zhang, J.-J. Shan, Y. Zhu, A. I. Frenkel, A. Patlolla, W. Huang, S. J. Yoon, L. Wang, H. Yoshida and S. Takeda, *J. Am. Chem. Soc.*, 2013, **135**, 8283–8293.
- 152 A. Grimaud, O. Diaz-Morales, B. Han, W. T. Hong, Y.-L. Lee, L. Giordano, K. A. Stoerzinger, M. T. Koper and Y. Shao-Horn, *Nat. Chem.*, 2017, **9**, 457–465.
- 153 T. Tang, W.-J. Jiang, S. Niu, N. Liu, H. Luo, Y.-Y. Chen, S.-F. Jin, F. Gao, L.-J. Wan and J.-S. Hu, *J. Am. Chem. Soc.*, 2017, **139**, 8320–8328.
- 154 Y. Lin, L. Yang, H. Jiang, Y. Zhang, Y. Bo, P. Liu, S. Chen, B. Xiang, G. Li and J. Jiang, *J. Phys. Chem. Lett.*, 2020, **11**, 1746–1752.
- 155 C. Zhao, N. Li, R. Zhang, Z. Zhu, J. Lin, K. Zhang and C. Zhao, *ACS Appl. Mater. Interfaces*, 2019, **11**, 47858–47867.
- 156 M. Li, H. Liu and L. Feng, *Electrochem. Commun.*, 2021, **122**, 106901.
- 157 S. Chen, S. Zhang, L. Guo, L. Pan, C. Shi, X. Zhang, Z.-F. Huang, G. Yang and J.-J. Zou, *Nat. Commun.*, 2023, **14**, 4127.
- 158 Y.-P. Deng, Y. Jiang, R. Liang, S.-J. Zhang, D. Luo, Y. Hu, X. Wang, J.-T. Li, A. Yu and Z. Chen, *Nat. Commun.*, 2020, **11**, 1952.
- 159 Y. Zhao, N. Dongfang, C. A. Triana, C. Huang, R. Erni, W. Wan, J. Li, D. Stoian, L. Pan and P. Zhang, *Energy Environ. Sci.*, 2022, **15**, 727–739.
- 160 C. Kaithwas, P. Bhuyan, S. Pradhan and S. Mandal, *Mater. Charact.*, 2018, **145**, 582–593.
- 161 C. Athreya, K. Deepak, D.-I. Kim, B. De Boer, S. Mandal and V. S. Sarma, *J. Alloys Compd.*, 2019, **778**, 224–233.
- 162 E. Bousquet, M. Dawber, N. Stucki, C. Lichtensteiger, P. Hermet, S. Gariglio, J.-M. Triscone and P. Ghosez, *Nature*, 2008, **452**, 732–736.
- 163 S. Farokhipoor, C. Magén, S. Venkatesan, J. Íñiguez, C. J. Daumont, D. Rubi, E. Snoeck, M. Mostovoy, C. De Graaf and A. Müller, *Nature*, 2014, **515**, 379–383.
- 164 A. Yadav, C. Nelson, S. Hsu, Z. Hong, J. Clarkson, C. Schlepütz, A. Damodaran, P. Shafer, E. Arenholz and L. Dedon, *Nature*, 2016, **530**, 198–201.



- 165 J.-H. Kang, L. Xie, Y. Wang, H. Lee, N. Campbell, J. Jiang, P. J. Ryan, D. J. Keavney, J.-W. Lee and T. H. Kim, *Nano Lett.*, 2018, **18**, 6347–6352.
- 166 S. Cheng, C. Xu, S. Deng, M.-G. Han, S. Bao, J. Ma, C. Nan, W. Duan, L. Bellaiche and Y. Zhu, *Sci. Adv.*, 2018, **4**, eaar4298.
- 167 H.-Y. Sun, G.-R. Xu, F.-M. Li, Q.-L. Hong, P.-J. Jin, P. Chen and Y. Chen, *J. Energy Chem.*, 2020, **47**, 234–240.
- 168 Y. Liu, J. Wang, S. Kim, H. Sun, F. Yang, Z. Fang, N. Tamura, R. Zhang, X. Song and J. Wen, *Nature*, 2019, **570**, 358–362.
- 169 H. Wang, H. Zhang, Y. Huang, H. Wang, A. Ozden, K. Yao, H. Li, Q. Guo, Y. Liu and A. Vomiero, *ACS Nano*, 2022, **17**, 346–354.
- 170 Z. Yang, D. Ji, Z. Li, Z. He, Y. Hu, J. Yin, Y. Hou, P. Xi and C. H. Yan, *Small*, 2023, **19**, 2303099.
- 171 X. Long, S. Xiao, Z. Wang, X. Zheng and S. Yang, *Chem. Commun.*, 2015, **51**, 1120–1123.
- 172 Z. Chen, M. Ju, M. Sun, L. Jin, R. Cai, Z. Wang, L. Dong, L. Peng, X. Long and B. Huang, *Angew. Chem., Int. Ed.*, 2021, **60**, 9699–9705.
- 173 C. M. Gabardo, A. Seifitokaldani, J. P. Edwards, C.-T. Dinh, T. Burdyny, M. G. Kibria, C. P. O'Brien, E. H. Sargent and D. Sinton, *Energy Environ. Sci.*, 2018, **11**, 2531–2539.
- 174 X. Wang, C. Liu, C. Gao, K. Yao, S. S. M. Masouleh, R. Berté, H. Ren, L. D. S. Menezes, E. Cortés and I. C. Bicket, *ACS Nano*, 2021, **15**, 10553–10564.
- 175 Z. Chen, X. Zhang and G. Lu, *Chem. Sci.*, 2015, **6**, 6829–6835.
- 176 J.-W. Zhao, Z.-X. Shi, C.-F. Li, Q. Ren and G.-R. Li, *ACS Mater. Lett.*, 2021, **3**, 721–737.
- 177 H. J. Song, H. Yoon, B. Ju and D. W. Kim, *Adv. Energy Mater.*, 2021, **11**, 2002428.
- 178 Z. Lu, H. Wang, D. Kong, K. Yan, P.-C. Hsu, G. Zheng, H. Yao, Z. Liang, X. Sun and Y. Cui, *Nat. Commun.*, 2014, **5**, 4345.
- 179 Y. Li, X. Wu, J. Wang, H. Wei, S. Zhang, S. Zhu, Z. Li, S. Wu, H. Jiang and Y. Liang, *Electrochim. Acta*, 2021, **390**, 138833.
- 180 M. Yu, J. Li, F. Liu, J. Liu, W. Xu, H. Hu, X. Chen, W. Wang and F. Cheng, *J. Energy Chem.*, 2022, **72**, 361–369.
- 181 W. Liu, J. Yu, M. G. Sendeku, T. Li, W. Gao, G. Yang, Y. Kuang and X. Sun, *Angew. Chem., Int. Ed.*, 2023, **135**, e202309882.
- 182 L. Trotochaud, S. L. Young, J. K. Ranney and S. W. Boettcher, *J. Am. Chem. Soc.*, 2014, **136**, 6744–6753.
- 183 J. S. Kim, I. Park, E. S. Jeong, K. Jin, W. M. Seong, G. Yoon, H. Kim, B. Kim, K. T. Nam and K. Kang, *Adv. Mater.*, 2017, **29**, 1606893.
- 184 H. Liang, A. N. Gandhi, C. Xia, M. N. Hedhili, D. H. Anjum, U. Schwingenschlogl and H. N. Alshareef, *ACS Energy Lett.*, 2017, **2**, 1035–1042.
- 185 F. Dong, H. Duan, Z. Lin, H. Yuan, M. Ju, X. Du, J. Gao, J. Yu and S. Yang, *Appl. Catal., B*, 2024, **340**, 123242.
- 186 M. Görlin, P. Chernev, J. Ferreira de Araújo, T. Reier, S. R. Dresch, B. Paul, R. Krähnert, H. Dau and P. Strasser, *J. Am. Chem. Soc.*, 2016, **138**, 5603–5614.
- 187 N. Li, D. K. Bediako, R. G. Hadt, D. Hayes, T. J. Kempa, F. Von Cube, D. C. Bell, L. X. Chen and D. G. Nocera, *Proc. Natl. Acad. Sci. U. S. A.*, 2017, **114**, 1486–1491.
- 188 S. Lee, A. Moysiadou, Y.-C. Chu, H. M. Chen and X. Hu, *Energy Environ. Sci.*, 2022, **15**, 206–214.
- 189 R. Mehmood, W. Fan, X. Hu, J. Li, P. Liu, Y. Zhang, Z. Zhou, J. Wang, M. Liu and F. Zhang, *J. Am. Chem. Soc.*, 2023, **145**, 12206–12213.
- 190 C. Feng, X. She, Y. Xiao and Y. Li, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218738.
- 191 J. Kang, X. Qiu, Q. Hu, J. Zhong, X. Gao, R. Huang, C. Wan, L.-M. Liu, X. Duan and L. Guo, *Nat. Catal.*, 2021, **4**, 1050–1058.
- 192 D. Wang, W. Yan, S. H. Vijapur and G. G. Botte, *J. Power Sources*, 2012, **217**, 498–502.
- 193 N. Chen, Y.-X. Du, G. Zhang, W.-T. Lu and F.-F. Cao, *Nano Energy*, 2021, **81**, 105605.
- 194 A. S. Aricò, V. Baglio and V. Antonucci, *Electrocatal. Direct Methanol Fuel Cells*, 2009, 1–78.
- 195 X. Bai, X. Zhao, Y. Zhang, C. Ling, Y. Zhou, J. Wang and Y. Liu, *J. Am. Chem. Soc.*, 2022, **144**, 17140–17148.
- 196 X. Wang, M. Xie, F. Lyu, Y.-M. Yiu, Z. Wang, J. Chen, L.-Y. Chang, Y. Xia, Q. Zhong and M. Chu, *Nano Lett.*, 2020, **20**, 7751–7759.
- 197 J. Bau, S. Kozlov and L. Azofra, *J. Mater. Chem. A*, 2019, **7**, 15031–15035.
- 198 R. Cai, M. Sun, J. Ren, M. Ju, X. Long, B. Huang and S. Yang, *Chem. Sci.*, 2021, **12**, 15382–15388.
- 199 R. Cai, M. Sun, F. Yang, M. Ju, Y. Chen, M. D. Gu, B. Huang and S. Yang, *Chemistry*, 2024, **10**, 211–233.
- 200 C. Zhu, H. Li, S. Fu, D. Du and Y. Lin, *Chem. Soc. Rev.*, 2016, **45**, 517–531.
- 201 J. Masa, W. Xia, M. Muhler and W. Schuhmann, *Angew. Chem., Int. Ed.*, 2015, **54**, 10102–10120.
- 202 H. T. Chung, D. A. Cullen, D. Higgins, B. T. Sneed, E. F. Holby, K. L. More and P. Zelenay, *Science*, 2017, **357**, 479–484.
- 203 G. Wu, K. L. More, C. M. Johnston and P. Zelenay, *Science*, 2011, **332**, 443–447.
- 204 K. Gong, F. Du, Z. Xia, M. Durstock and L. Dai, *Science*, 2009, **323**, 760–764.
- 205 S. Anantharaj, S. R. Ede, K. Sakthikumar, K. Karthick, S. Mishra and S. Kundu, *ACS Catal.*, 2016, **6**, 8069–8097.
- 206 Q. Ma, C. Hu, K. Liu, S.-F. Hung, D. Ou, H. M. Chen, G. Fu and N. Zheng, *Nano Energy*, 2017, **41**, 148–153.
- 207 V. S. Saji and C. W. Lee, *ChemSusChem*, 2012, **5**, 1146–1161.
- 208 L. Su, D. Han, G. Zhu, H. Xu, W. Luo, L. Wang, W. Jiang, A. Dong and J. Yang, *Nano Lett.*, 2019, **19**, 5423–5430.
- 209 C. Xie, Z. Niu, D. Kim, M. Li and P. Yang, *Chem. Rev.*, 2019, **120**, 1184–1249.
- 210 C. Zhu, Z. Zhang, L. Zhong, C.-S. Hsu, X. Xu, Y. Li, S. Zhao, S. Chen, J. Yu and S. Chen, *Chemistry*, 2021, **7**, 406–420.
- 211 Y. Wang, J. Liu and G. Zheng, *Adv. Mater.*, 2021, **33**, 2005798.



- 212 K. Jiang, R. B. Sandberg, A. J. Akey, X. Liu, D. C. Bell, J. K. Nørskov, K. Chan and H. Wang, *Nat. Catal.*, 2018, **1**, 111–119.
- 213 S. Martens, L. Asen, G. Ercolano, F. Dionigi, C. Zalis, A. Hawkins, A. M. Bonastre, L. Seidl, A. C. Knoll and J. Sharman, *J. Power Sources*, 2018, **392**, 274–284.
- 214 S. M. Alia, S. Shulda, C. Ngo, S. Pylypenko and B. S. Pivovar, *ACS Catal.*, 2018, **8**, 2111–2120.
- 215 H. A. El-Sayed, A. Weiß, L. F. Olbrich, G. P. Putro and H. A. Gasteiger, *J. Electrochem. Soc.*, 2019, **166**, F458–F464.
- 216 P. Lettenmeier, R. Wang, R. Abouatallah, S. Helmlj, T. Morawietz, R. Hiesgen, S. Kolb, F. Burggraf, J. Kallo and A. S. Gago, *Electrochim. Acta*, 2016, **210**, 502–511.
- 217 Y. Peng, K. Jiang, W. Hill, Z. Lu, H. Yao and H. Wang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 3971–3977.
- 218 D. Li, E. J. Park, W. Zhu, Q. Shi, Y. Zhou, H. Tian, Y. Lin, A. Serov, B. Zulevi and E. D. Baca, *Nat. Energy*, 2020, **5**, 378–385.
- 219 F.-Y. Chen, Z.-Y. Wu, Z. Adler and H. Wang, *Joule*, 2021, **5**, 1704–1731.
- 220 M. Chandesris, V. Médeau, N. Guillet, S. Chelghoum, D. Thoby and F. Fouda-Onana, *Int. J. Hydrogen Energy*, 2015, **40**, 1353–1366.
- 221 E. J. Park, S. Maurya, M. R. Hibbs, C. H. Fujimoto, K.-D. Kreuer and Y. S. Kim, *Macromolecules*, 2019, **52**, 5419–5428.
- 222 P. Thangavel, M. Ha, S. Kumaraguru, A. Meena, A. N. Singh, A. M. Harzandi and K. S. Kim, *Energy Environ. Sci.*, 2020, **13**, 3447–3458.
- 223 M. Bernt and H. A. Gasteiger, *J. Electrochem. Soc.*, 2016, **163**, F3179.
- 224 H.-S. Oh, H. N. Nong, T. Reier, A. Bergmann, M. Gliech, J. Ferreira de Araújo, E. Willinger, R. Schlögl, D. Teschner and P. Strasser, *J. Am. Chem. Soc.*, 2016, **138**, 12552–12563.
- 225 S. Geiger, O. Kasian, A. M. Mingers, S. S. Nicley, K. Haenen, K. J. Mayrhofer and S. Cherevko, *ChemSusChem*, 2017, **10**, 4140–4143.
- 226 M. Görlin, J. Ferreira de Araújo, H. Schmies, D. Bernsmeier, S. R. Dresp, M. Gliech, Z. Jusys, P. Chernev, R. Kraehnert and H. Dau, *J. Am. Chem. Soc.*, 2017, **139**, 2070–2082.
- 227 J. P. Meyers and R. M. Darling, *J. Electrochem. Soc.*, 2006, **153**, A1432.
- 228 Q. Dong, T. Li, Y. Yao, X. Wang, S. He, J. Li, J. Luo, H. Zhang, Y. Pei and C. Zheng, *Joule*, 2020, **4**, 2374–2386.
- 229 A. Li, H. Ooka, N. Bonnet, T. Hayashi, Y. Sun, Q. Jiang, C. Li, H. Han and R. Nakamura, *Angew. Chem., Int. Ed.*, 2019, **58**, 5054–5058.
- 230 S. Z. Oener, M. J. Foster and S. W. Boettcher, *Science*, 2020, **369**, 1099–1103.
- 231 X. Zhao, H. Ren and L. Luo, *Langmuir*, 2019, **35**, 5392–5408.
- 232 B. Han, K. A. Stoerzinger, V. Tileli, A. D. Gamalski, E. A. Stach and Y. Shao-Horn, *Nat. Mater.*, 2017, **16**, 121–126.
- 233 W. Tong, M. Forster, F. Dionigi, S. Dresp, R. Sadeghi Erami, P. Strasser, A. J. Cowan and P. Farràs, *Nat. Energy*, 2020, **5**, 367–377.
- 234 S. Dresp, T. N. Thanh, M. Klingenhof, S. Brückner, P. Hauke and P. Strasser, *Energy Environ. Sci.*, 2020, **13**, 1725–1729.
- 235 L. Chen, J.-T. Ren and Z.-Y. Yuan, *Sci. China Mater.*, 2022, **65**, 2433–2444.
- 236 L. Chen, J.-T. Ren and Z.-Y. Yuan, *Green Chem.*, 2022, **24**, 713–747.
- 237 L. Piccolo, P. Afanasiev, F. Morfin, T. Len, C. Dessal, J. L. Rousset, M. Aouine, F. Bourgain, A. Aguilar-Tapia and O. Proux, *ACS Catal.*, 2020, **10**, 12696–12705.
- 238 J. Yi, Z. Zhou, Y. Xia, G. Zhou, G. Zhang, L. Li, X. Wang, X. Zhu, X. Wang and H. Pang, *Chin. Chem. Lett.*, 2023, **34**, 108328.
- 239 H. Zhang, S. Zuo, M. Qiu, S. Wang, Y. Zhang, J. Zhang and X. W. Lou, *Sci. Adv.*, 2020, **6**, eabb9823.
- 240 X. Tao, W. Shi, B. Zeng, Y. Zhao, N. Ta, S. Wang, A. A. Adenle, R. Li and C. Li, *ACS Catal.*, 2020, **10**, 5941–5948.
- 241 B. Yan, J. Li, Z. Lin, C. Du and G. Yang, *ACS Appl. Nano Mater.*, 2019, **2**, 6783–6792.
- 242 S. Wu, C. Li, Y. Wang, Y. Zhuang, Y. Pan, N. Wen, S. Wang, Z. Zhang, Z. Ding and R. Yuan, *Angew. Chem., Int. Ed.*, 2023, **135**, e202309026.
- 243 Y. Wang, Y. Ma, X.-B. Li, L. Gao, X.-Y. Gao, X.-Z. Wei, L.-P. Zhang, C.-H. Tung, L. Qiao and L.-Z. Wu, *J. Am. Chem. Soc.*, 2020, **142**, 4680–4689.
- 244 J. Fan, Y. Zhao, H. Du, L. Zheng, M. Gao, D. Li and J. Feng, *ACS Appl. Mater. Interfaces*, 2022, **14**, 26752–26765.
- 245 S. Xie, Y. Li, B. Sheng, W. Zhang, W. Wang, C. Chen, J. Li, H. Sheng and J. Zhao, *Appl. Catal., B*, 2022, **310**, 121320.
- 246 Y. Zhang, Z. Xu, Q. Wang, W. Hao, X. Zhai, X. Fei, X. Huang and Y. Bi, *Appl. Catal., B*, 2021, **299**, 120679.
- 247 Y. Dou, A. Zhou, Y. Yao, S. Y. Lim, J.-R. Li and W. Zhang, *Appl. Catal., B*, 2021, **286**, 119876.
- 248 F. Fang, Y. Liu, X. Sun, C. Fu, Y. P. Bhoi, W. Xiong and W. Huang, *Appl. Surf. Sci.*, 2021, **564**, 150407.
- 249 R. Li, Q. Luan, C. Dong, W. Dong, W. Tang, G. Wang and Y. Lu, *Appl. Catal., B*, 2021, **286**, 119832.
- 250 Z. Pászti, S. Bálint, P. Németh, G. P. Szijjártó, A. Tompos and E. Tálas, *Mater. Res. Bull.*, 2016, **83**, 65–76.
- 251 X. Zheng, D. Liu, J. Wen and S. Lv, *Sep. Purif. Technol.*, 2021, **275**, 117934.
- 252 J. Prince, F. Tzompantzi, G. Mendoza-Damián, F. Hernández-Beltrán and J. S. Valente, *Appl. Catal., B*, 2015, **163**, 352–360.
- 253 M. Suárez-Quezada, G. Romero-Ortiz, V. Suárez, G. Morales-Mendoza, L. Lartundo-Rojas, E. Navarro-Cerón, F. Tzompantzi, S. Robles, R. Gómez and A. Mantilla, *Catal. Today*, 2016, **271**, 213–219.
- 254 Y. Lu, W.-J. Yin, K.-L. Peng, K. Wang, Q. Hu, A. Selloni, F.-R. Chen, L.-M. Liu and M.-L. Sui, *Nat. Commun.*, 2018, **9**, 2752.
- 255 K. Zhao, W. Pang, S. Jiang, C. Hu, P. Liu, D. Cui, X. An, B. Tian, C. Gao and P. Zhang, *Nano Res.*, 2023, **16**, 4812–4820.



- 256 S. Xie, C. Deng, Q. Huang, C. Zhang, C. Chen, J. Zhao and H. Sheng, *Angew. Chem., Int. Ed.*, 2023, **62**, e202216717.
- 257 C. Y. Toe, Z. Zheng, H. Wu, J. Scott, R. Amal and Y. H. Ng, *Angew. Chem., Int. Ed.*, 2018, **57**, 13613–13617.
- 258 L. Chen, J. T. Ren and Z. Y. Yuan, *Adv. Energy Mater.*, 2023, **13**, 2203720.
- 259 L. Chen, X.-L. Song, J.-T. Ren and Z.-Y. Yuan, *Appl. Catal., B*, 2022, **315**, 121546.
- 260 H. Huang, H. Song, J. Kou, C. Lu and J. Ye, *J. Energy Chem.*, 2022, **67**, 309–341.
- 261 J. Li, W. Pan, Q. Liu, Z. Chen, Z. Chen, X. Feng and H. Chen, *J. Am. Chem. Soc.*, 2021, **143**, 6551–6559.

