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Random lasing realized in *n*-ZnO/*p*-MgZnO core-shell nanowire heterostructures

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Well-aligned ZnO nanowire arrays have been prepared, and *p*-MgZnO has been deposited onto the nanowires to form core-shell heterostructures. Transmission electron microscopy confirms the formation of *n*-ZnO/*p*-MgZnO core-shell nanowire heterostructures. Under the injection of continuous current, random lasing with a threshold current of around 15 mA has been observed from the heterostructures. The low threshold may be due to the relatively high crystalline quality of the ZnO nanowires as well as the carrier confinement in the heterostructures.

Introduction

Random lasing is the lasing phenomenon realized in random media, and it has attracted much attention in recent years for its significance in both fundamental research and potential applications in versatile fields, such as displaying, sensing biological imaging, etc.[1-6] Many efforts have been devoted to develop random lasers, and optically pumped random lasers have been reported extensively.[7-13] Realizing electrically pumped lasing has been one of the most challenging issues of random lasers, and zinc oxide (ZnO) has been considered as one of the most active materials for electrically pumped random lasers.[14-19] Although many reports have demonstrated electrically pumped random lasing from ZnO based films, most of them are realized in metal-insulation-semiconductor (MIS) structures.[15,20-24] It is accepted that in MIS structures, holes are generated via an impact ionization process under relatively high bias voltage. However, the generation efficiency of holes via impact ionization process is very low. Additionally, the relatively high bias voltage may cause serious heating effect, thus degrades the performance of the random lasers greatly. A more efficient way for the generation and injection of holes is to construct *p*-*n* junctions, in which electrons and holes can be injected from the *n*-type and *p*-type layer into the active layer and recombine radiatively there.[25,26] However, very few reports on ZnO-based electrically pumped random lasing from *p*-*n* homojunctions can be found up to date, possibly due to the reproducible *p*-type doping of ZnO-based materials is still a challenging issue.[18,27-29] If an *n*-ZnO/*p*-MgZnO heterostructure can be constructed, in which the MgZnO has a wider bandgap, holes can then be injected into ZnO, and recombine with the electrons there, thus electrically pumped random lasing may be realized. However, none such report can be found up to date.

By employing a lithium nitrogen codoping method, we have realized reproducible *p*-type ZnO films, and reliable light-emitting devices that can work continuously for several hours have been demonstrated.[30,31] In this paper, *n*-ZnO/*p*-MgZnO core-shell heterostructures have been fabricated by employing ZnO nanowires as the active layer. In these structures the high crystalline and optical quality of the nanowires may facilitate the occurrence of lasing, and the relatively large surface to volume ratio of the nanowires will provide an arena for strong scattering that is necessary for random lasing. Under forward bias, electrically pumped random lasers have been realized, and the threshold is around 15 mA, which is much smaller than the corresponding value realized in ZnO *p*-*n* junctions (~40 mA).

Experimental

The ZnO nanowires were grown on *a*-plane sapphire substrates by metal-organic chemical vapor deposition (MOCVD) technique. The precursors used for the growth were diethylzinc and oxygen, and high-purity (9N) nitrogen was used as the carrier gas to lead the precursors into the growth chamber. Prior to the growth, the substrates were treated at 850°C under 10^{-4} Pa for 60 minutes to remove the possible absorbed contaminants. During the growth process, the substrate temperature was kept at 650°C, and the pressure in the MOCVD chamber was kept at 3×10^{3} Pa. After two hours growth, the nanowire samples were loaded into a plasma-assisted molecular beam epitaxy facility (VG V80H) for the deposition of *p*-type MgZnO layer to form the *n*-ZnO/*p*-MgZnO core-shell nanowire heterostructures. The detailed growth conditions for the Li and N codoped *p*-type MgZnO layer can be found elsewhere.[31-33] Briefly, nitric oxide (NO) and oxygen (O_2) cracked via an Oxford Applied Research HD25 radio-frequency (13.56 MHz) atomic source were used as the N dopant and O source, respectively.

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Metallic zinc, magnesium and lithium contained in individual Knudsen effusion cells were used as Zn, Mg and Li sources. During the growth process, the pressure in the growth chamber was fixed at 2×10^{-5} mbar, and the NO and O_2 flow rate were maintained at 0.9 and 1.0 sccm, respectively. The substrate temperature was kept at 650 °C. Metallic Ni/Au and In layer were deposited onto the *p*-type MgZnO layers and ZnO nanowires by thermal evaporation method to form the ohmic contacts. The thickness of Ni/Au electrode is about 30 nm/500 nm. The morphology of the ZnO nanowires was characterized using a Hitachi S4800 field-emission scanning electron microscope (SEM). The crystalline properties of the ZnO nanowires and core-shell structures were evaluated by a Bruker D8 Discover x-ray diffractometer (XRD) with Cu Kα (λ= 1.54 Å) as the radiation source and a JEM-2010 transmission electron microscope (TEM). Electrical properties of the MgZnO layer were measured by a Hall system (LakeShore 7707) under van der Pauw configuration. Electroluminescence (EL) measurements of the heterostructures were carried out in a Hitachi F4500 spectrometer employing a continuous-wave current as the excitation source.

Results and discussion

Figures 1(a) and 1(b) show the plan-view and side-view SEM images of the ZnO nanowires that act as the active layer of the random lasers. One can see that well-aligned nanowires have been grown vertically on the sapphire substrate. The length of the nanowires is around 3.8 μm, and the nanowires taper in diameter from around 150 nm at the base to around 40 nm at the top. The XRD pattern of the ZnO nanowire arrays is shown in Fig. 1(c). Only a peak at 34.4° is visible from the pattern besides the diffraction from the sapphire substrate, which can be indexed to the diffraction from the (0002) facet of wurtzite ZnO. The dominant (0002) diffraction peaks reveals that the nanowires may be grown along the [0001] direction of wurtzite structure. The x-ray rocking curve of the nanowires is illustrated in the inset of Fig. 1(c), from which a Gaussian peak with a full width at half maximum (FWHM) of around 0.46° can be observed. The Phi-scan spectrum of the nanowires is shown in Fig. 1(d), and six peaks with equal interval of 60° can be defined from the spectrum, indicating the six-fold symmetry of the nanowires.

A typical TEM image of the ZnO nanowires is shown in Fig. 2(a), and the nanowires show smooth surface, revealing that each nanowire is well crystallized. Also one can see the tapered structure of the nanowires clearly from the image. The high-resolution TEM image of an individual ZnO nanowire is shown in Fig. 2(b). Well-defined lattice fringes can be observed, indicating the high crystalline quality of the nanowires. The spacing between two adjacent lattice fringes is around 0.51 nm, which corresponds to the *d*-spacing of (0001) plane in hexagonal ZnO. From the lattice fringes, one can also see that the nanowire grown along [0001] direction, which is consistent with the XRD result shown in Fig. 1(c). Figure 2(c) shows the selected area electron diffraction (SAED) pattern of an individual nanowire. The clear dotted pattern confirms the single-crystalline nature of the nanowires.

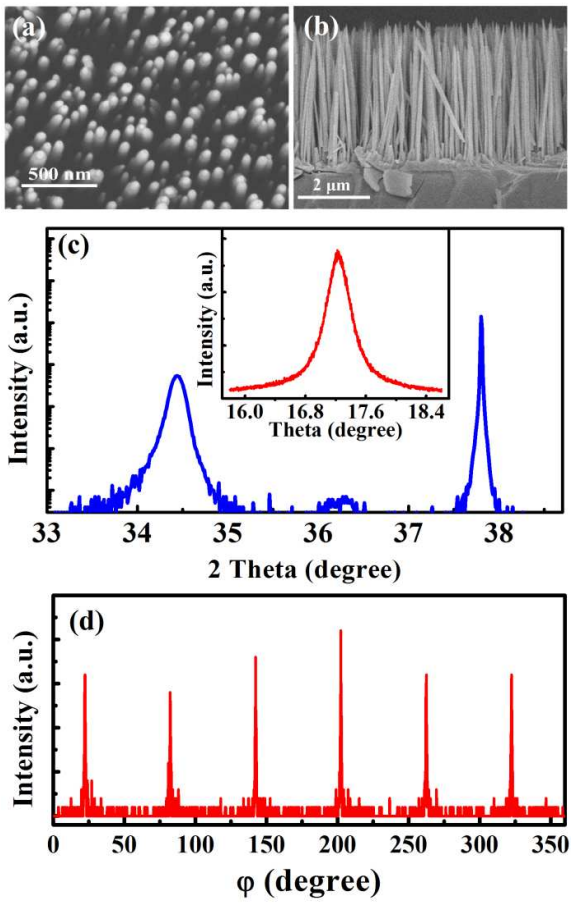

Fig. 1 Plan-view (a) and side-view (b) SEM images of the ZnO nanowires; (c) θ-2θ XRD pattern of the ZnO nanowires, and the inset shows the rocking curve of the nanowires; (d) x-ray phi-scan spectrum of the nanowires.

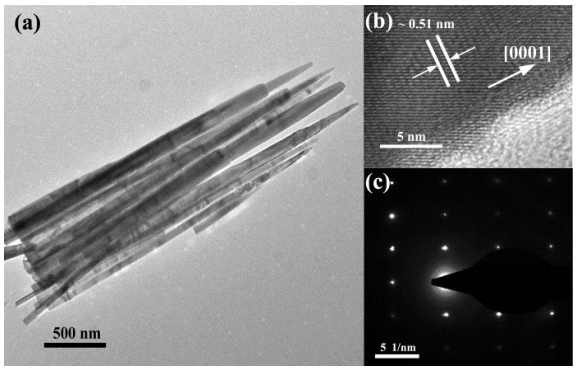

Fig. 2 TEM image of the ZnO nanowires (a); the high-resolution TEM image (b) and selected area electron diffraction pattern (c) of an individual ZnO nanowire.

To achieve electrically pumped random lasing, *p*-type MgZnO films have been deposited onto the nanowires to form *n*-ZnO/*p*-MgZnO core-shell nanowires heterostructures. The Mg content in the MgZnO layer determined by energy dispersive x-ray spectroscopy (EDS) is 0.35. The electrical properties of the $Mg_{0.35}Zn_{0.65}O$ layers on the nanowires cannot be measured directly by Hall measurement because it will be interfered greatly by the nanowires. Thus to assess the electrical properties of the $Mg_{0.35}Zn_{0.65}O$ layers, Hall measurements have been carried out on

the layers grown directly on sapphire substrate in the same growth process. The layer shows *p*-type conduction with a hole

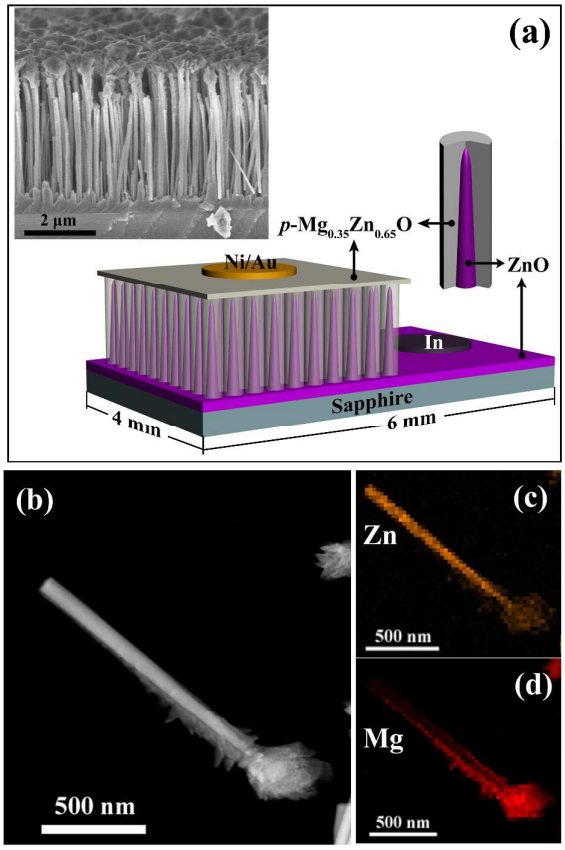

Fig. 3 (a) Schematic diagram of the *n*-ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ core-shell nanowire heterostructure, and the inset shows the side-view SEM image of the device; (b) Dark field Scanning TEM image of an individual ZnO/MgZnO core-shell nanowire heterostructure, and the EDS elemental mapping of Zn (c) and Mg (d).

concentration of around 9×10^{16} cm^{-3} and a Hall mobility of 0.8 $cm^2V^{-1}s^{-1}$. It is then deduced that the $Mg_{0.35}Zn_{0.65}O$ layers deposited onto the ZnO nanowires are *p*-type conduction. The schematic illustration of the device is shown in Fig. 3(a). The size of the device is 4 mm× 6 mm, as marked in the diagram. The inset of Fig. 3(a) shows the side-view SEM image of the *n*-ZnO/*p*-MgZnO core-shell nanowires structure. One can see clearly from the image that the nanowires are grown vertically onto the sapphire substrate with an ZnO layer lying underneath the nanowires. Also there is a thin MgZnO layer on the top of the nanowires. Figure 3(b) shows a typical dark field scanning transmission electron microscope image of an individual *n*-ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ core-shell nanowire heterostructure. One can see from the image that the ZnO nanowire has been coated with a thin $Mg_{0.35}Zn_{0.65}O$ layer to form the core-shell structure. The thickness of the MgZnO shell around the top of the nanowires is about 90 nm. The Zn and Mg elemental mapping of the core-shell structure is shown in Figs. 3(c) and 3(d). These figures reveal that Zn is mainly concentrated in the core area, while Mg atoms are only distributed within the shell area, which clearly reveals the formation of core-shell heterostructures.

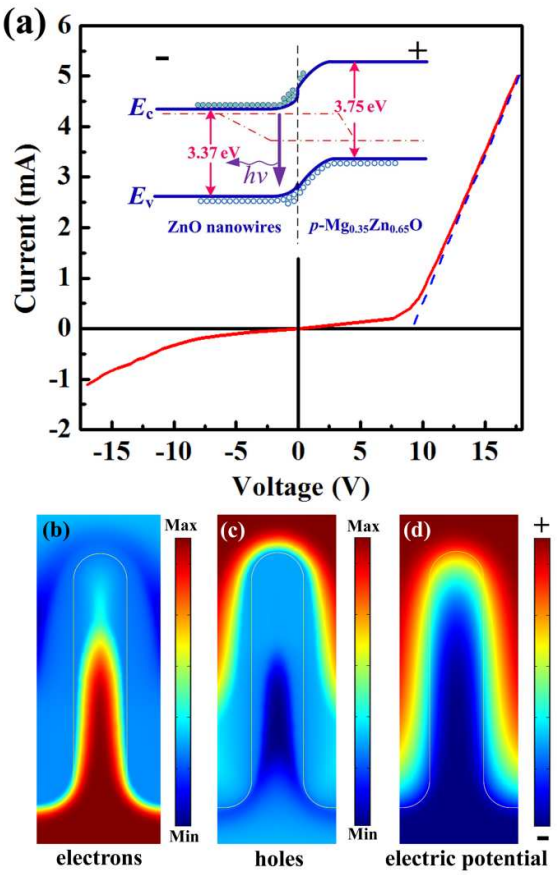

Fig. 4 (a) *I– V* curve of the *p-n* heterostructure, and the inset shows the bandgap diagram under forward bias; (b) the electrons distribution, (c) the holes distribution, (d) the electric potential distribution in the core-shell nanowire heterostructure.

The current-voltage (*I- V*) characteristic of the heterostructure shows obvious rectification behaviors with a turn-on voltage of around 9.0 V, as shown in Fig. 4(a). The bandgap diagram of the heterostructure is shown in the inset of Fig. 4(a). Under forward bias, holes in the *p*-$Mg_{0.35}Zn_{0.65}O$ layer will be drifted into the ZnO nanowires, while the electrons will be confined into the ZnO nanowires because of the barrier formed by the conduction band offset at the ZnO/$Mg_{0.35}Zn_{0.65}O$ interface. The distribution of electrons and holes in the *n*-ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ core-shell heterostructure has been simulated using finite-difference time-domain (FDTD) method, as shown in Figs. 4(b) and 4(c). One can find that electrons are mainly confined in the ZnO nanowires, and holes in the *p*-$Mg_{0.35}Zn_{0.65}O$ shell layer surround the nanowires, and some of them can be injected into the nanowires. Then the electrons confined in the nanowires will recombine with the injected holes from the *p*-$Mg_{0.35}Zn_{0.65}O$ shell layer to produce photons. Figure 4(d) illustrates the distribution of the electric potential in the core-shell heterostructure. The electric potential variation occurs mainly at the ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ interface, which reveals that the electrons and holes will mainly recombine at the interface of the core-shell heterostructure. The relatively high crystalline quality of the nanowires may facilitate the occurrence of lasing, while the large surface of the nanowires may provide an arena for strong scattering that is necessary for random lasing.

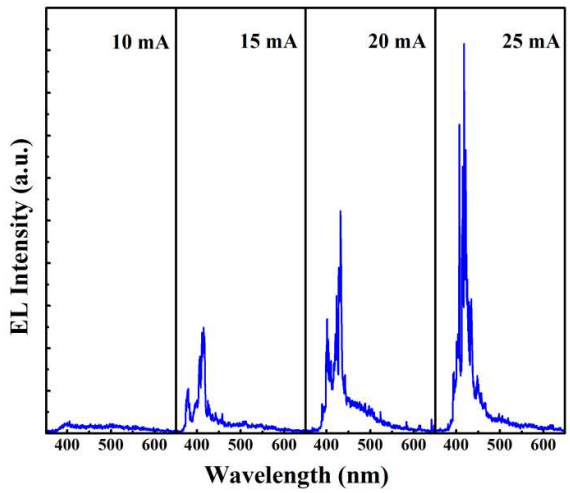

Fig. 5 Room temperature EL spectra of the heterostructure at different injection current.

To test the above speculation, a forward bias has been applied onto the *n*-ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ structures, and obvious emission has been detected, the emission spectrum of which is illustrated in Fig. 5. When the injection current is 10 mA, a weak broad emission can be detected. The spectrum is composed of two bands, and the one at 398 nm can be attributed to the near band-edge (NBE) emission of ZnO, while the other band at around 500 nm are usually attributed to the deep-level emission in ZnO.[34,35] When the injection current is increased to 15 mA, the intensity of the NBE emission has been enhanced significantly, and on the broad spontaneous emission. When the injection current is increased further, the intensity of the sharp peaks increases greatly, and the FWHM of the sharp peaks is around 1.3 nm. The above phenomenon has been frequently observed in ZnO MIS or *p-n* junction structures,[15,18,23,28] and has been attributed to the occurrence of electrically pumped random lasers. A proof supports the formation of random lasing is that the emission shows multidirectional distribution, as indicated in Fig. 6. One can see that the spectra collected at different observation angles show different lasing peaks, which is typical characteristic for a random laser.[18,36] Note that although there have been some reports on ultraviolet emission in ZnO core-shell nanowire structures,[37-40] none report on electrically pumped random laser in ZnO core-shell nanowire *p-n* heterostructures can be found before.

The mechanism for the random laser observed in our case can be understood as follows: According to the carrier and electric potential distribution shown in Fig. 4, the recombination of electrons and holes will occur at the interface of the ZnO/$Mg_{0.35}Zn_{0.65}O$ interface, and photons are mainly produced there. The photons will undergo multiple scattering when they come out of the nanowires. Under the effect of the multiple scattering, photons may return to the nanowires where they are emitted, thus close-loops will be formed. When the injection current is small, the optical gain in the close-loops is smaller than the loss, thus only broad spontaneous emission can be observed from the structure, as the situation shown in the 10 mA spectrum in Fig. 5. While at larger injection current, the gain in the close-loops may exceed the loss, thus lasing will be observed, as indicated by the spectra with injection current of 15 mA and above in Fig. 5. Since the close-loops are formed randomly between the nanowires, thus the emission recorded from different angles shows different lasing spectra. Note that the threshold of the electrically pumped random lasers (around 15 mA) is smaller than the corresponding values ever reported in Au/SiO_x/ZnO or Au/MgO/ZnO MIS structures (usually around 70 mA)[15,22-24] and *p-n* homojunctions (~ 40 mA)[18], and the low threshold may originate from the relatively high crystalline quality of the nanowires and the confinement of the carriers in the heterostructures.

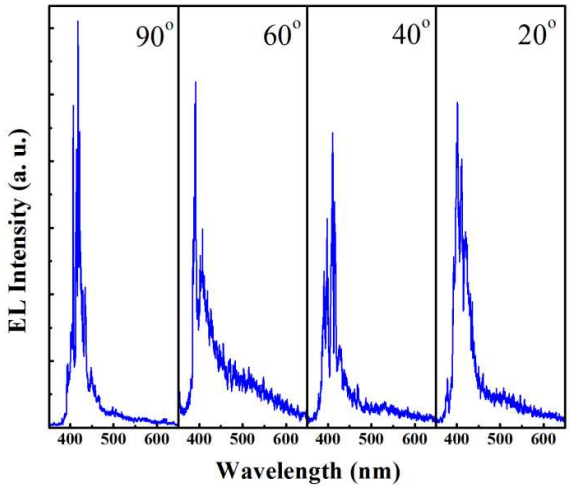

Fig. 6 Angular-dependent EL spectra of the device under the injection current of 25 mA, in which 90° means the detector is perpendicular to the surface of the sample.

Conclusions

In conclusion, *n*-ZnO/*p*-$Mg_{0.35}Zn_{0.65}O$ core-shell nanowire heterostructures have been fabricated, and random lasing has been observed from the structure under the injection of continuous current. A threshold of around 15 mA has been achieved, which is smaller than the corresponding values ever reported in similar MIS and *p-n* junction structures, and the small threshold has been attributed to the relatively high crystalline quality of the nanowires, as well as the carriers confinement in the heterostructures. The results reported in this paper reveal that nanowire core-shell *p-n* heterostructures may be a promising structure for electrically pumped random lasers.

Acknowledgment

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Graphical Abstract

Random lasing realized in *n*-ZnO/*p*-MgZnO core-shell nanowire heterostructures

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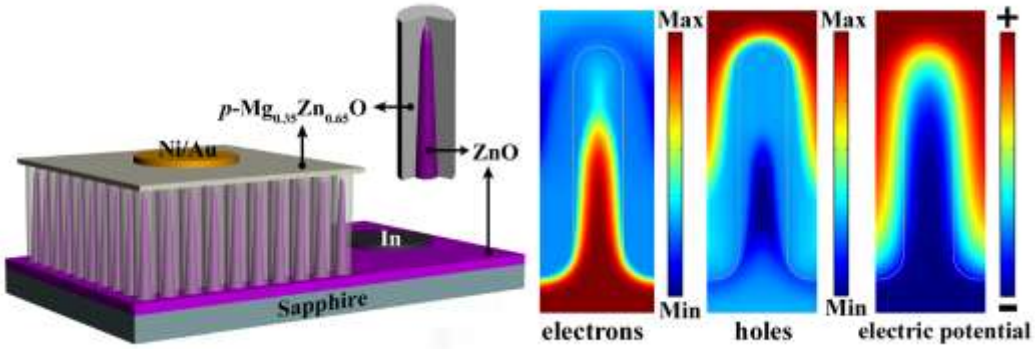

Low threshold electrically pumped random lasers were realized in *p*-MgZnO/*n*-ZnO core-shell nanowire heterostructures.