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An aromatic diamine-constructed lead-free polar perovskite towards stable self-driven X-ray detection

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Zero-dimensional (0D) bismuth halide perovskites are attractive candidate semiconductors for X-ray detection due to their low biotoxicity and inhibited ionic migration. However, achieving stable X-ray detection applications of 0D perovskites remains challenging. Introducing aromatic diamine cations into perovskites can enhance structural stability and improve charge transport behaviors. Herein, utilizing the strong anchoring effect of the diamine with the inorganic skeleton, stable X-ray detection is achieved by synthesizing novel 0D polar perovskites, namely AP_2BiI_7 (AP = 4-amidinopyridine). In particular, leveraging intrinsic spontaneous electric polarization that produces a significant 1.15 V photovoltage, the device achieves excellent self-driven X-ray detection at zero electric field, with a high sensitivity of 58 μC Gy^{-1} cm^{-2} and a low detection limit of 100 nGy s^{-1}. Furthermore, the detector exhibits low baseline drift and highly stable X-ray detection performance, benefiting from the anchoring effect of the aromatic diamine cations in the compound. Our work represents a significant step toward realizing stable, eco-friendly, self-powered X-ray detectors based on perovskite materials.

Introduction

Direct X-ray detection technology converts X-ray photons directly into electrical signals and is widely used in medical imaging, scientific research, industrial inspection, and other fields.[1–5] Halide perovskites, with their high mobility–lifetime product, strong X-ray absorption, and simple synthesis, show great promise for direct X-ray detection due to their excellent optoelectronic properties.[6–9] $MAPbI_3$-based (MA = methylamine) direct X-ray detectors, for instance, achieve sensitivities as high as 2.2×10^8 μC Gy^{-1} cm^{-2} and detection limits down to 0.1 nGy s^{-1}.[10,11] However, the device instability due to severe ionic migration remains a significant challenge for the real application of perovskite detectors.[12–14] Furthermore, equipment instability could lead to toxic lead leaking from Pb-based perovskites, resulting in significant contamination of soil and groundwater resources.[15,16] Therefore, developing "green" lead-free perovskites for stable X-ray detection is urgent.

Bismuth ions (Bi^{3+}) share similarities in electronic structure, ionic radius, and electronegativity with lead ions (Pb^{2+}). Despite the difference in charge, these similarities make bismuth a promising, less-toxic alternative to lead in many applications.[17,18] Bismuth halide perovskites (BHPs) have garnered attention for their large resistivity and low ionic mobility, properties associated with enhanced stability.[19,20] For example, Liu and colleagues synthesised a classical 0D BHP, $MA_3Bi_2I_9$, from aliphatic amine cations.[12] Aromatic amine cations are more polar than their aliphatic counterparts, which enhances electronic coupling between inorganic octahedra. This reduces quantum confinement and improves charge transport.[21] More recently, aromatic diamine-derived BHPs can further effectively enhance the stability of materials and inhibit ionic migration due to the anchoring effect of aromatic diamine cations.[22–24] Yang *et al.* constructed the BHP (3-(aminomethyl)pyridine)BiI_5 using aromatic diamine cations. The material achieved efficient charge transfer and suppressed ion migration, which ultimately led to stable photodetection performance.[25] However, most currently known BHPs constructed from aromatic diamines lack polar space group symmetry in

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their crystalline structure. This necessitates the application of an external electric field to achieve efficient charge carrier separation and transport,[12,26] resulting in complex device architectures and significant energy consumption.[27] Therefore, achieving stable self-powered X-ray detection in BHPs constructed with aromatic diamines is essential.

In this work, by introducing aromatic diamine AP^{2+} (AP = 4-amidinopyridine), we have successfully synthesized new polar 0D BHPs, AP_2BiI_7. Anchoring aromatic diamines *via* both termini to an inorganic framework enhances stability and charge carrier mobility in self-driven detection. Moreover, the device based on polar AP_2BiI_7 exhibits a strong bulk photovoltaic effect (BPVE) with a photovoltage of 1.15 V due to the excellent spontaneous polarization exhibited along the polar *c*-axis of 35.53 μC cm^{-2}. Under zero applied electric field utilizing the BPVE, the detector demonstrates a high sensitivity (58 μC Gy^{-1} cm^{-2}) and a low detection limit (100 nGy s^{-1}). Moreover, the aromatic diamine cation promotes the stability of the AP_2BiI_7 structure; thus, the detector shows a low I_{drift} of 3.8×10^{-7} nA cm^{-1} s^{-1} V^{-1} at 500 V cm^{-1} and demonstrates excellent environmental stability. Our work provides new insights into exploring novel 0D bismuth halide perovskite materials for stable self-driven X-ray detection.

Results and discussion

Single-crystal X-ray diffraction (SCXRD) at 300 K confirmed the structure of AP_2BiI_7, which reveals that AP_2BiI_7 adopts the *Cc* polar space group (a = 14.2687 Å, b = 14.3448 Å, c = 15.7364 Å) (Table S1). The polar structure exhibits a significant spontaneous polarization along the *c*-axis direction, with calculated polarization values from the point charge model reaching 35.53 μC cm^{-2} (Fig. S1 and Table S2). It is therefore promising to observe the obvious bulk photovoltaic effect in the *c*-axis direction. The second harmonic generation signal of AP_2BiI_7 powders is measured to be about 0.11 times that of KH_2PO_4 (KDP), further confirming the non-centrosymmetric structure of AP_2BiI_7 (Fig. S2). The structure contains the inorganic $[BiI_6]^{3-}$ octahedron and isolated I^- anion, which are charge-balanced by two independent aromatic diamine AP^{2+} cations, forming a 0D structure (Fig. 1a). The Bi–I bond lengths in the inorganic skeleton range from 3.0249(10) Å to 3.1590(10) Å, and the I–Bi–I angles range from 83.12(3)° to 97.18(3)° (Tables S3 and S4).

The degree of octahedral distortion of AP_2BiI_7 was quantified using the following equation to calculate the distortion index (Δd) and the bond angle variance (σ^2):[28]

$$\Delta d = \frac{1}{n}\sum_{i=1}^{n}\left(\frac{d_i - d}{d}\right)^2 \quad (1)$$

$$\sigma^2 = \frac{1}{11}\sum_{n=1}^{12}(\theta_n - 90°)^2 \quad (2)$$

in which the variable d_i (or d) is the individual (or mean) Bi–I bond length, while θ_n denotes the bond angle between neighboring Bi–Br bonds. $\Delta d = 1.64 \times 10^{-4}$ and σ^2 = 20.4 deg^2 have

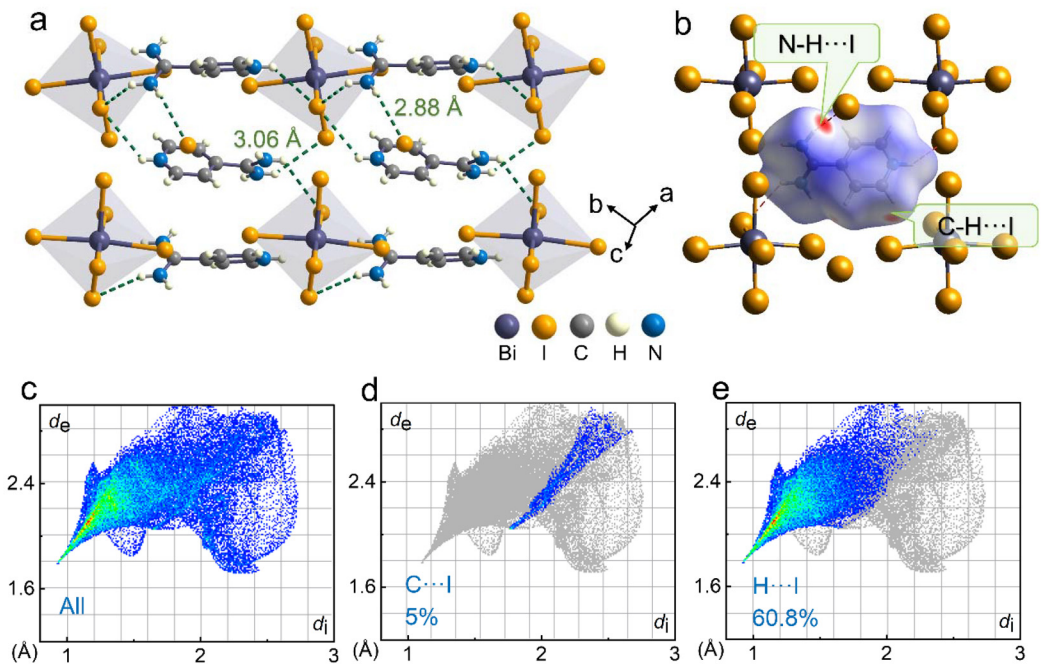

Fig. 1 (a) The crystal structure of AP_2BiI_7, in which N–H···I hydrogen bonds between aromatic diamine AP^{2+} cations and inorganic skeletons are connected by dotted lines. (b) The Hirshfeld d_{norm} surfaces (since the distance between the atoms is less than the sum of their van der Waals radii, the red regions on the Hirshfeld surface indicate a closer interaction between them) of the aromatic diamine AP^{2+} and (c–e) the corresponding 2D fingerprint plots for C···I and H···I contacts in the AP_2BiI_7 structure.

been calculated. This indicates that the AP_2BiI_7 SC structure has minor structural distortion, which facilitates carrier transport. Furthermore, the AP^{2+} molecule bridges and stabilizes the inorganic framework *via* hydrogen bonds, shortening inter-octahedra distances and suppressing ion migration. The distance between neighboring inorganic skeletons is shorter than that constructed from monoamines. The shortest I–I distance between adjacent $[BiI_6]^{3-}$ octahedra of AP_2BiI_7 is 4.072 Å, which is shorter than those in $(PBA)_4BiI_7{\cdot}H_2O$ (6.998 Å, PBA = $C_6H_5(CH_2)_4NH_3^+$),[29] $MA_3Bi_2I_9$ (4.57 Å), and BZA_3BiI_6 (4.195 Å, BZA = benzylamine).[30] Furthermore, as demonstrated by the Hirshfeld surface analysis, there is a notable molecular electrostatic interaction between the electron-rich I^- anions and the benzene ring ($I^-\cdots\pi$). This interaction is attributed to the proximate distance between the I^- anion and the neighboring benzene ring, which is approximately 3.7 Å (Fig. 1b–d and Fig. S3).

The Hirshfeld surface analysis reveals that strong N–H···I hydrogen bonds between the AP^{2+} cations and the inorganic framework account for 60.8% of the total interactions. This indicates strong hydrogen bonding between aromatic diamine cations (AP^{2+}) and the inorganic skeleton of AP_2BiI_7. The stronger hydrogen bonding reduces the distance of adjacent $[BiI_6]^{3-}$-octahedra and promotes lattice rigidity and charge transport, resulting in good photoelectric properties.

Red SCs of AP_2BiI_7 with around $5.2 \times 4 \times 2$ mm^3 were grown from a stoichiometric reaction of Bi_2O_3 with AP in a hot hydriodic acid solution *via* the slow temperature-cooling process (Fig. 2a). PXRD patterns confirm their phase purity (Fig. S4). Meanwhile, the morphology of AP_2BiI_7 was simulated using Mercury software, which matched the obtained bulk SCs and determined the *c*-axis orientation (Fig. S5). Additionally, a d_{33} value of 4.9 pC N^{-1} characterizes the clear piezoelectric response observed along the *c*-axis in AP_2BiI_7 crystals (Fig. S6).

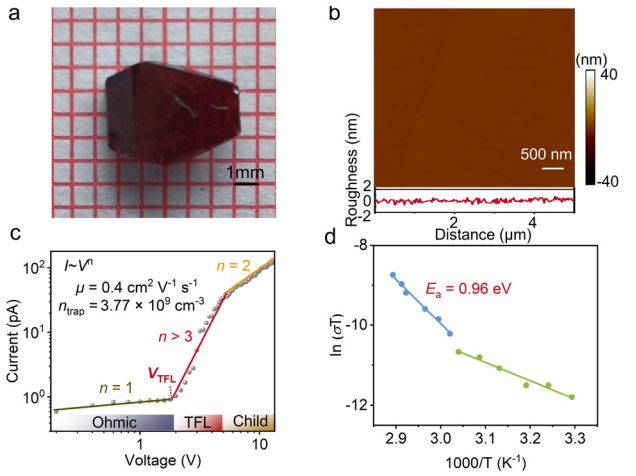

Fig. 2 (a) An image of a SC of AP_2BiI_7. (b) The AFM image of the AP_2BiI_7 crystal. (c) The *I*–*V* curve of the AP_2BiI_7 crystal was obtained using the SCLC method. (d) The conductivity of AP_2BiI_7 is temperature-dependent.

Thus, it is further verified that AP_2BiI_7 is polar. Furthermore, the AFM image of the AP_2BiI_7 crystal surface revealed that it was very smooth and flat with few imperfections. The mean roughness and root-mean-square roughness were found to be 0.235 and 0.304 nm, respectively, thereby confirming the exceptional quality of the crystals, referring to Fig. 2b. The high quality of AP_2BiI_7 SCs is conducive to excellent carrier transport. Internal crystal defects were further assessed by measuring trap density (n_{trap}) in AP_2BiI_7 SCs with the space-charge-limited current (SCLC) technique.

The current–voltage (*I*–*V*) curve is subdivided into three distinct regions, as illustrated in Fig. 2c. These are Trap Filling Limited (TFL) ($n > 3$), Ohmic ($n = 1$), and Child ($n = 2$) regions. The n_{trap} value is calculated using the following equation:[31]

$$n_{trap} = \frac{2\varepsilon\varepsilon_0 V_{TFL}}{eL^2} \quad (3)$$

where V_{TFL} is the threshold voltage, ε_0 represents the vacuum dielectric constant, e denotes the unit charge, ε designates the relative dielectric constant, and the variable L indicates the length of the conductive channel. The n_{trap} of AP_2BiI_7 is calculated as 3.77×10^9 cm^{-3}. The value is comparable to those of high-quality $MA_3Bi_2I_9$ (1.2×10^{10} cm^{-3})[12] and (3-AMP)BiI_5 (3.53×10^9 cm^{-3}, 3-AMP = 3-(aminomethyl)pyridine)[32] (for full data, see Table S5). The low n_{trap} will favor charge transport and thus improve the mobility–lifetime ($\mu\tau$) product of AP_2BiI_7. Moreover, the hole-only carrier mobility (μ), determined from the Child region using the Mott–Gurney power law, is:[33,34]

$$\mu = \frac{8JL^3}{9\varepsilon\varepsilon_0 V^2}$$

where J, L and V refer to the dark current density, thickness and applied voltage, respectively. The μ value of AP_2BiI_7 was calculated to be 0.4 cm^2 V^{-1} s^{-1}, which is higher than those of bismuth-based perovskites such as $Cs_2AgBiBr_6$ (0.05 cm^2 V^{-1} s^{-1})[35] and $FA_3Bi_2I_9$ (0.0089 cm^2 V^{-1} s^{-1}),[26] indicating that AP_2BiI_7 has excellent charge transport properties. Furthermore, the bulk resistivity (ρ) of AP_2BiI_7 SCs is found to be 3.55×10^{10} Ω cm by fitting the current–voltage curve along the *c*-axis (Fig. S7), which is comparable to those of $MA_3Bi_2I_9$ (3.74×10^{10} Ω cm),[12] $FA_3Bi_2I_9$ (7.8×10^{10} Ω cm, FA = formamidinium),[26] and $AG_3Bi_2I_9$ (3.78×10^{10} Ω cm, AG = aminoguanidinium).[36] Therefore, AP_2BiI_7 with large resistivity is more capable of suppressing dark current, which contributes to improving X-ray detection performance. We tested the absorption spectra of AP_2BiI_7 to illustrate its optical properties (Fig. S8). The absorption edge of the material is located at ~646 nm, and the optical band gap was further calculated to be ~2.0 eV by fitting the Tauc equation. Then, we performed density-functional theory (DFT) calculations on AP_2BiI_7 (Fig. S9),[37–39] and the results show that the material is characterized by an indirect bandgap. The calculated band gap is 1.928 eV, which is in good agreement with the experimental measurements (2.0 eV) and verifies the accuracy of the calculated model. The partial density of states profiles show that the conduction band minimum (CBM) is mainly contributed

by N-2p and C-2p orbitals, while the valence band maximum (VBM) is dominated by I-4p orbitals with contributions from Bi-6p orbitals. Combined with the charge density distributions at the VBM and CBM, it can be confirmed that the CBM of the material is dominated by its organic cation, while the VBM is mainly determined by the iodide ions in the inorganic components. To further understand the effect of aromatic diamines on the inhibition of ion migration (as shown in Fig. 2d), the activation energy can be obtained from the Nernst–Einstein equation, see the SI for more details. The E_a value of this compound is 0.96 eV. This is much larger than those of $FA_3Bi_2I_9$ (0.56 eV)[26] and (TMBD)$BiBr_5$ (0.64 eV, *N*,*N*,*N*′,*N*′-tetramethyl-1,4-butanediammonium).[40] This implies that the strong contact between the diamine cations and the inorganic skeletons is useful in increasing the energy barrier for ion migration, and the ion migration is effectively suppressed.

The X-ray performance of the AP_2BiI_7 SC was investigated by fabricating it into SC devices. The X-ray device type with an Ag/AP_2BiI_7 SC/Ag two-terminal structure was adopted; see Fig. 3a and Fig. S10. Through the photon cross-section database,[41] we calculated the X-ray absorption spectra of $MA_3Bi_2I_9$, α-Se, Si, and AP_2BiI_7 (Fig. 3b). AP_2BiI_7 demonstrates a markedly greater linear absorption coefficient compared to Si. This value approaches that of α-Se, indicating effective X-ray absorption across a wide photon energy range (1 to 1000 keV) for AP_2BiI_7. Fig. 3c shows the attenuation efficiency of AP_2BiI_7 for 50 keV X-ray photons. AP_2BiI_7 with a thickness of 1 mm can absorb approximately 94% of incident photons, which is considerably higher than the ratio (≈9.7%) observed for Si. Excellent X-ray attenuation efficiency and efficient charge collection (quantified by $\mu\tau$) are both essential for high-performance detectors. Here, the voltage-dependent photocurrent of

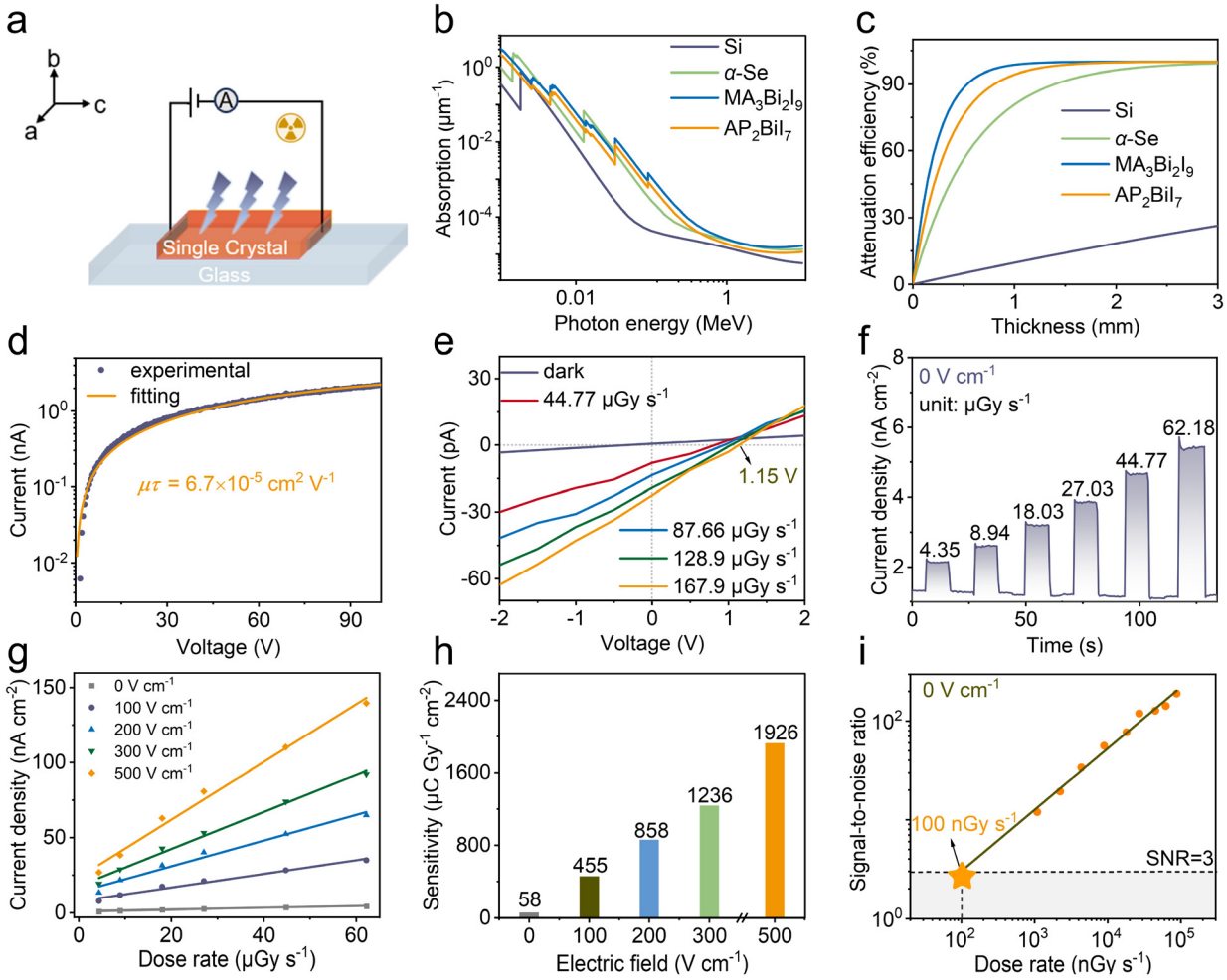

Fig. 3 (a) Diagram of the AP_2BiI_7 SC device along the polar *c*-axis direction. (b) AP_2BiI_7, Si, α-Se, and $MA_3Bi_2I_9$ absorption coefficients as a function of photon energies. (c) AP_2BiI_7, Si, α-Se, and $MA_3Bi_2I_9$ attenuation efficiencies of 50 keV X-ray photons as a function of thickness. (d) The voltage-dependent photocurrent of the AP_2BiI_7 detector yielded a $\mu\tau$ value of 6.7 × 10^{-5} cm^2 V^{-1} under X-ray irradiation. (e) *I*–*V* traces of the AP_2BiI_7 SC detector in the dark and when exposed to X-ray radiation. (f) The current densities of the AP_2BiI_7 SC detector at 0 V cm^{-1} with increased dose rates. (g) The X-ray-induced photocurrent density depends on the dose rate. By fitting the slopes, the sensitivity of the AP_2BiI_7 detector can be determined. (h) The sensitivity of the AP_2BiI_7 detector under different electric fields. (i) The X-ray dose rate-dependent SNR for the AP_2BiI_7 SC detector at 0 V cm^{-1}.

AP_2BiI_7 SC detectors can be fitted with the modified Hecht equation to determine the $\mu\tau$ value.

$$I = \frac{I_0 \mu\tau V}{L^2}\left[-\exp\left(-\frac{L^2}{\mu\tau V}\right)\right] \quad (4)$$

where L and V represent the electrode spacing and the applied voltage, and I_0 is the saturated photocurrent. The $\mu\tau$ value for AP_2BiI_7, measured at 6.7×10^{-5} cm^2 V^{-1} (Fig. 3d), matches the range observed in some BHPs, such as $(R\text{-PPA})_2BiI_5$ (5.6×10^{-5} cm^2 V^{-1}, R-PPA = R-1-phenylpropylamine)[41] and $(BAH)BiI_4$ (1.95×10^{-4} cm^2 V^{-1}, BAH = benzamidinium),[42] and $(FA)_3Bi_2I_9$ (2.4×10^{-5} cm^2 V^{-1}).[26] The AP_2BiI_7 detector has enormous potential for X-ray detection, benefiting from its high resistivity, efficient charge collection, and outstanding X-ray absorption. The I–V traces of the AP_2BiI_7 detector along the c-axis are depicted in Fig. 3e, both in the dark and at various X-ray dosage rates. We can observe a significant BPVE of 1.15 V. The BPVE originates from the inherent spontaneous polarization of polar crystals. The device is capable of driving the separation and subsequent transport of photogenerated carriers, thereby conferring upon AP_2BiI_7 the ability to perform self-driven detection. Indeed, even at 0 V cm^{-1}, the AP_2BiI_7 detector shows excellent photoresponse (Fig. 3f).

Its exceptional response to X-rays is demonstrated by the current density growing linearly from 4.35 to 62.18 μGy s^{-1} as the X-ray dosage rate increases. For comparison, tests were also performed at 100, 200, 300, and 500 V cm^{-1} electric fields (Fig. S11). As expected, as the external voltage increases, it is more favorable for carrier collection and therefore higher current densities can be obtained. For AP_2BiI_7 SC detectors, sensitivity (S) is a key performance index, which can be determined using the following formula:[43]

$$S = \frac{(J_{ph} - J_d)}{D} \quad (5)$$

where D is the X-ray dose rate, J_{ph} represents the photocurrent density, and J_d denotes the dark current density. Fig. 3g shows that the photocurrent density minus the dark current density (J_{ph} – J_d) is linearly associated with the X-ray dosage rate. By fitting their slopes, the corresponding sensitivities at different electric fields were calculated; see Fig. 3h. Even in the self-driven mode, the sensitivity of the AP_2BiI_7 SC detector (58 μC Gy^{-1} cm^{-2}) exceeds those of the commercial α-Se film detector (20 μC Gy^{-1} cm^{-2}@10 V μm^{-1}) and 0D monovalent cation BHPs of $(R\text{-PPA})_2BiI_5$ (31 μC Gy^{-1} cm^{-2}@0 V).[41] Its sensitivity reaches 1926 μC Gy^{-1} cm^{-2} at 500 V cm^{-1} as the electric field gradually increases. The value exceeds those of many reported BHP X-ray detectors (for full data, see Table S6).[44]

Furthermore, it should be noted that as the applied electric field increases, the dark current density also increases. For example, the dark current density at 500 V cm^{-1} is about 37 times that at 0 V cm^{-1}. Self-powered mode reduces the dark current, allowing for a lower detection limit. By the definition set forth by the International Union of Pure and Applied Chemistry (IUPAC), the limit of detection (LoD) is defined as a signal-to-noise ratio (SNR) of 3, used to evaluate the detection performance of analyti-

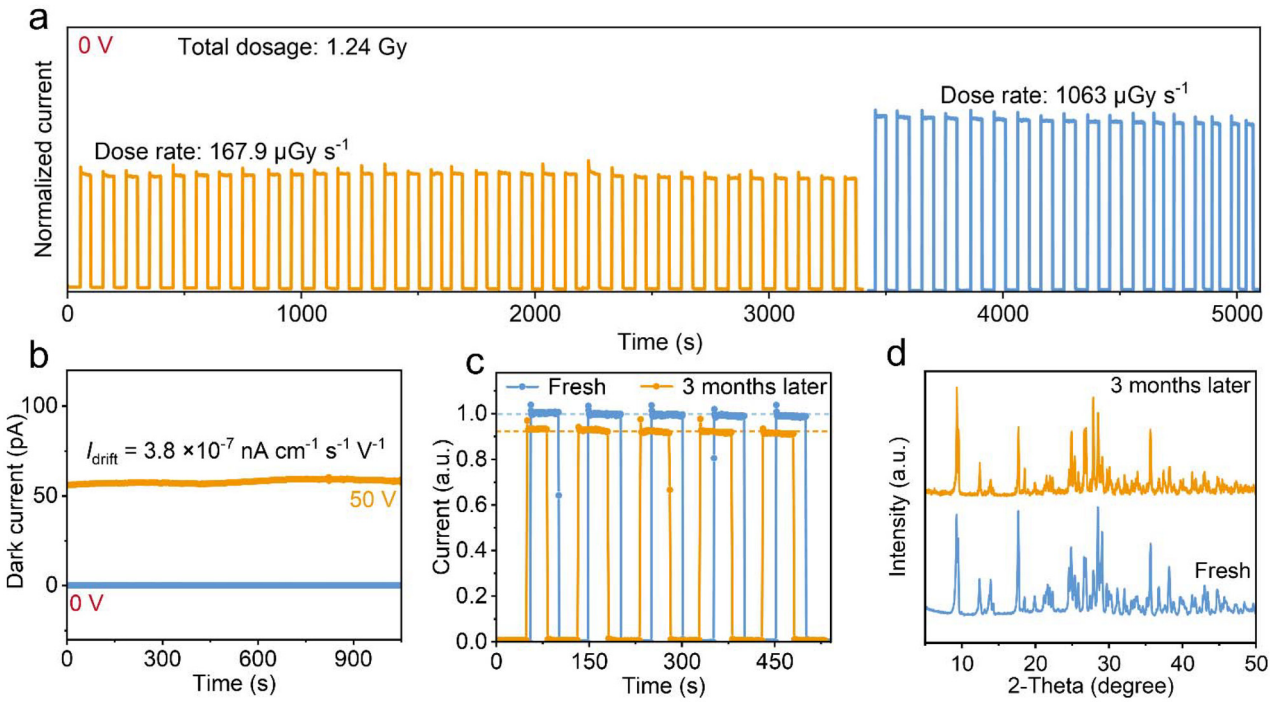

Fig. 4 (a) X-ray stability test of the unencapsulated AP_2BiI_7 detectors exposed to X-ray irradiation for more than an hour of continuous operation. (b) The dark current was measured. (c) Initial radiation photoresponse of the AP_2BiI_7 detector at a high dose rate of 1063 μGy s^{-1} and 0 V cm^{-1}, and the radiation photoresponse after three months. (d) PXRD patterns of AP_2BiI_7 powders at preparation and after three months.

cal methods.[45] A LoD of 100 nGy s^{-1} at 0 V cm^{-1} was acquired by calculating the SNR at various dose rates, as seen in Fig. 3i, and then fitting the relationship between the SNRs and the dose rates. In addition, LoD values were calculated for different electric fields as shown in Fig. S12. These values significantly reduce the risk of X-ray injury, which is lower than the conventional medical diagnostic LoD of 5.5 μGy s^{-1}.[12,46]

The stability of equipment is another important performance indicator for its commercialization. The thermogravimetric decomposition temperature of AP_2BiI_7 can reach as high as 563 K, indicating its high thermal stability (Fig. S13). It is common to quantify this performance in X-ray detection in terms of radiation stability, I_{drift}, and environmental stability. To evaluate the radiation stability of the device, it was subjected to prolonged, high-dose-rate X-ray irradiation (167.9 μGy s^{-1} and 1063 μGy s^{-1}) at 0 V cm^{-1}. During 5100 s of continuous operation, the photocurrent of AP_2BiI_7 remained stable with a slight decline (Fig. 4a), demonstrating the detector's exceptional operational stability in the self-driven mode. In addition, the AP_2BiI_7 device exhibits excellent radiation stability even at 500 V cm^{-1} (Fig. S14). Furthermore, I_{drift} can be obtained from the following equation:[47]

$$I_{drift} = \frac{(J_t - J_0)}{Et} \quad (6)$$

where J_0 and J_t are the dark current density at the subsequent and initial times, respectively, while E represents the electric field. As is depicted in Fig. 4b, the AP_2BiI_7 SC detector not only obtains a markedly stable dark current at 0 V cm^{-1} but also demonstrates an I_{drift} value of 3.8×10^{-7} nA cm^{-1} s^{-1} V^{-1} for the AP_2BiI_7 SC device even under 500 V cm^{-1}. The value is considerably lower than those of previously reported BHP X-ray devices (Table S6).[48] This contrast again emphasizes the stability advantage of aromatic diamine structures. Finally, the stability in resisting ambient air of the device was investigated. As shown in Fig. 4c, after 90 days at a temperature of 24 ± 2 °C and a humidity of 52 ± 10%, the unencapsulated detector maintains a very high level of operational stability, with only a slight degradation of the X-ray photoresponse (approximately reduced to 92% of the original). In addition, the unencapsulated device retained approximately 90% of its sensitivity after 90 days (Fig. S15). The result demonstrates a level of environmental stability in the material system we developed. Furthermore, AP_2BiI_7 still maintains excellent phase stability by PXRD after three months (Fig. 4d). These indicate the superior environmental stability of the AP_2BiI_7 SC device. Such stability and exceptional self-driven X-ray detection performance exhibit the great commercial application prospect of AP_2BiI_7 perovskite devices.

Conclusions

In summary, a novel 0D lead-free polar perovskite, AP_2BiI_7, was synthesized by incorporating aromatic diamine AP^{2+} cations. High-quality SCs of AP_2BiI_7 exhibit a lower trap density n_{trap} (3.77×10^9 cm^{-3}) and a high resistivity (3.55×10^{10} cm^{-3}). Aromatic diamine AP^{2+} cations with an anchoring effect can effectively improve stability and charge transport behaviors. Moreover, such a polar structure thus gives them intrinsic spontaneous polarization along the polar axis direction, which further results in a strong BPVE with a photovoltage of 1.15 V under X-ray irradiation. It acts as the driving force for separating and transporting the X-ray-generated charge carriers, thereby endowing AP_2BiI_7 with the capability for self-driven detection. By leveraging the bulk photovoltage as the driving force, the 0D AP_2BiI_7 detector exhibits a notable sensitivity of 58 μC Gy^{-1} cm^{-2} and a low LoD of 100 nGy s^{-1} at 0 V cm^{-1}. Moreover, the aromatic diamine cation through the strong anchoring effect gives AP_2BiI_7 a stable structure, which makes AP_2BiI_7 show a low I_{drift} of 3.8×10^{-7} nA cm^{-1} s^{-1} V^{-1} at 500 V cm^{-1}, as well as excellent radiative stability both in self-driven mode and under applied electric fields. Impressively, even after three months, AP_2BiI_7 shows excellent environmental stability with only a slight decrease in photoresponse. This work demonstrates the first self-driven X-ray detection in 0D aromatic diamine-based BHPs, offering new insights for designing "green" and stable self-driven radiation detectors based on hybrid perovskites.

Author contributions

H. Zhong prepared the samples and wrote the manuscript. C. Qu investigated the photoelectric properties. S. You, Z. Wu, H. Ye, L. Xu, Y. Geng, H. Li, and C. Zhang provided suggestions for the project. Q. Guan and J. Luo designed and directed this project. All the authors discussed and commented on the manuscript. CRediT: Haiqing Zhong: data curation, formal analysis, investigation, visualization, and writing – original draft; Qianwen Guan: writing – review & editing; Shihai You: data curation and formal analysis; Zhenyue Wu: data curation and formal analysis; Huang Ye: data curation and formal analysis; Lijun Xu: data curation; Yaru Geng: data curation and formal analysis; Chengmin Ji: data curation; Hang Li: data curation; Chengshu Zhang: data curation; Chang Qu: data curation; Junhua Luo: formal analysis, funding acquisition, project administration, supervision, and writing – review & editing.

Conflicts of interest

There are no conflicts to declare.

Data availability

The data supporting this article have been included as part of the supplementary information (SI). Supplementary information: additional computational details and experimental details, materials, and methods, including crystal morphology,

crystal structure data, PXRD patterns, the TG curve, and basic photoelectric properties. See DOI: **https://doi.org/10.1039/d5qi01640a**.

CCDC 2401299 contains the supplementary crystallographic data for this paper.[49]

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