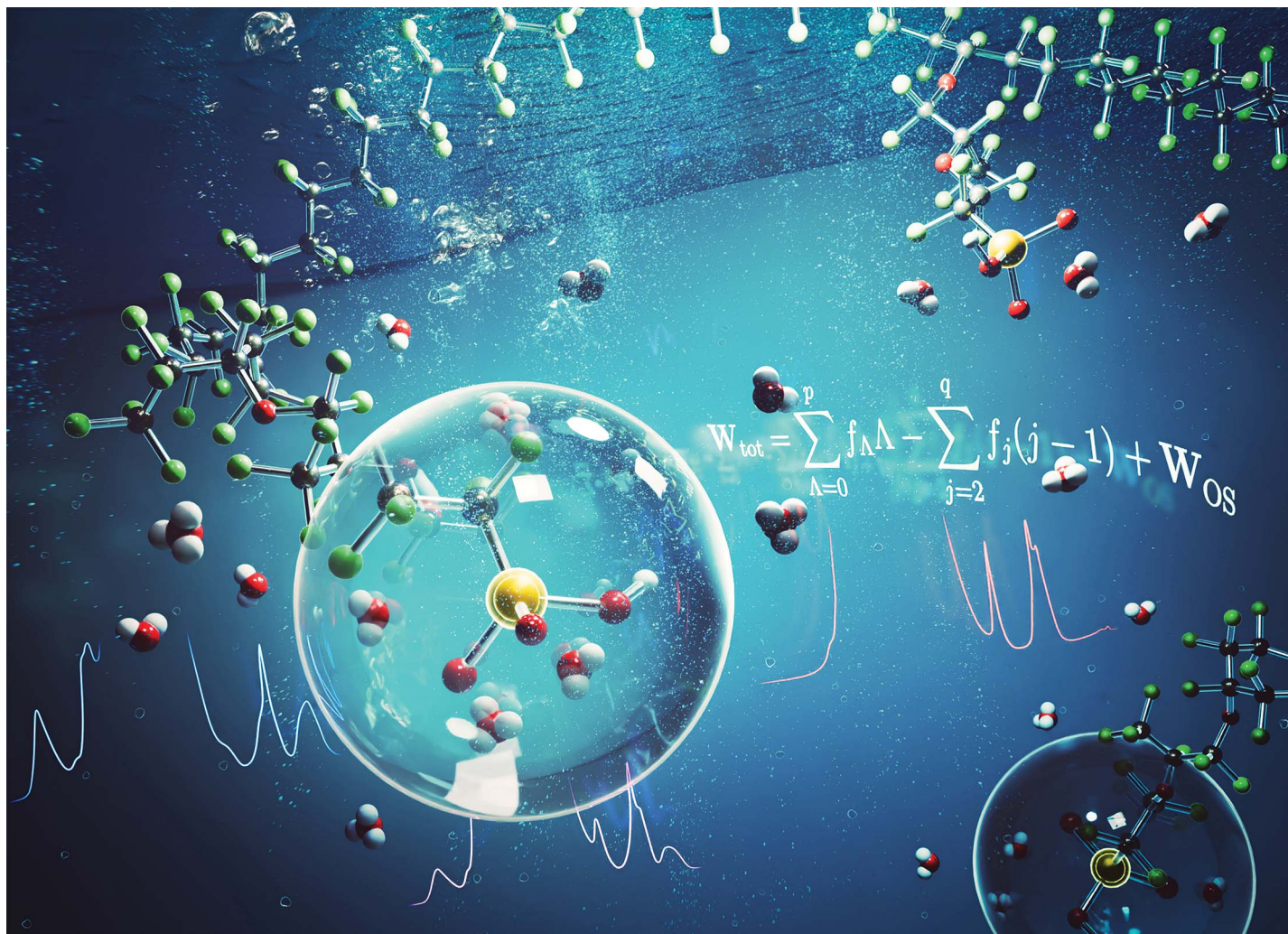




This collaborative study explores hydration effects in Nafion using simulations and spectroscopy, conducted by the labs of Dimakis (UTRGV), Jang (Georgia Tech), Goddard III (Caltech), and Smotkin (Northeastern).

Atomistic characterization of hydration-dependent fuel cell ionomer nanostructure: validation by vibrational spectroscopy

We use ReaxFF molecular dynamics and vibrational spectroscopy to study proton exchange in Nafion across nine hydration levels (λ). Simulations reveal that protonated exchange sites persist at high λ , challenging assumptions of complete ionization. By modeling realistic sites, we capture λ -dependent distributions of protonation states and local water content. These structures align with IR spectra and clarify ionomer hydration response.

Image reproduced by permission of Northeastern University from *J. Mater. Chem. A*, 2025, **13**, 24495

As featured in:



See Nicholas Dimakis,
William A. Goddard,
Eugene S. Smotkin *et al.*,
J. Mater. Chem. A, 2025, **13**, 24495.

Cite this: *J. Mater. Chem. A*, 2025, 13, 24495

Atomistic characterization of hydration-dependent fuel cell ionomer nanostructure: validation by vibrational spectroscopy†

Dan J. Donnelly III, ^{‡a} Moon Young Yang, ^{‡b} Nicholas Dimakis, ^{*c} Seung Soon Jang, ^d William A. Goddard III ^{*b} and Eugene S. Smotkin ^{*ae}

The development of Nafion alternatives for fuel cells and electrolyzers requires a fundamental understanding of hydration-dependent ion-exchange site acid/base chemistry. We present here reactive force field (ReaxFF) molecular dynamics (MD) simulations of Nafion at varying molar water/ion-exchange-site ratios (λ), which we correlate to our experimental and density functional theory-based vibrational spectra. ReaxFF describes the formation and breaking of covalent bonds, enabling simulations of proton exchange between sulfonic acid/sulfonate groups and water/hydronium. Our MD simulations determine the λ -dependent equilibrium proportions of protonated and deprotonated sites. We find that protonated sites persist across all λ from 0 to 20, challenging the widely accepted notion that all sites are ionized above a threshold λ value (e.g., 3 or 4). Our simulations generate hundreds of realistic exchange site environments, the characterization of which are based on 6 Å radii sulfur-centered 'inner-spheres'. These inner- and outer-sphere regions elucidate an interplay of stereoelectronic factors that influence protonation states, including the number of inner-sphere waters (Λ). Our simulations produce broad λ -dependent distributions of Λ values, representing non-uniform exchange site hydration. Moreover, we demonstrate that these distributions are specific to both protonated and deprotonated sites, with significant overlap between the two distributions for all non-zero λ . These distributions underpin the IR spectra of hydrated membranes, with each exchange site contributing an IR spectrum characteristic of its protonation state and Λ . We expect that these nanostructural characterizations of Nafion exchange sites will contribute to the development of new ionomers.

Received 16th May 2025

Accepted 3rd June 2025

DOI: 10.1039/d5ta03973h

rsc.li/materials-a

1 Introduction

Clean hydrogen technologies underpin long-term strategies established by the Paris Agreement.^{1,2} Efficacious use of clean hydrogen relies on infrastructure spanning production, storage, delivery, and consumption of the fuel which generates electricity and clean water as byproducts. The primary and terminal stages are accomplished by electrolyzers and fuel cells, respectively. Polymer electrolyte membrane (PEM) electrolyzers and

fuel cells are low-temperature (~ 80 °C) devices with over half a century of development.^{3–6} PEMs are solid-state ionomer membranes that also serve as separators of anode and cathode reactant streams, enabling compact system integration for rapid response to load demand changes.^{6–8}

Nafion is a benchmark perfluorinated sulfonic acid (PFSA) ionomer with superior chemical-mechanical stability and high protonic conductivity (Fig. 1).^{9–11} PFSA feature polytetrafluoroethylene (PTFE) backbones with randomly grafted perfluoroether sidechains, each terminated by a superacidic sulfonic acid group (*i.e.*, the proton exchange site). The PTFE backbone provides structural integrity and resistance to thermochemical degradation, while the exchange sites promote the sorption^{12–16} and retention of water required for proton transport *via* diffusion (vehicular) and Grotthuss (hopping or shuttling) mechanisms.^{17–21} The contrasting backbone and exchange site polarities give rise to a dynamic multiphase morphology driven by an interplay of electrostatic (hydration of ionic sites) and mechanical (deformation of backbone) force minimizations.^{10,14,22} Hydration dependent forces balance the distribution of hydrophobic (semicrystalline), hydrophilic (interphasial), and water-rich domains.^{9,10,23–25} Despite decades

^aDepartment of Chemistry and Chemical Biology, Northeastern University, Boston, MA 02115, USA. E-mail: e.smotkin@northeastern.edu^bMaterials and Process Simulation Center, California Institute of Technology, Pasadena, CA 91125, USA. E-mail: wag@caltech.edu^cDepartment of Physics and Astronomy, University of Texas Rio Grande Valley, Edinburg, TX 78539, USA. E-mail: nicholas.dimakis@utrgv.edu^dSchool of Materials Science and Engineering, Georgia Institute of Technology, Atlanta, GA 30332, USA^eDepartment of Chemical Engineering, Northeastern University, Boston, MA 02115, USA† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d5ta03973h>

‡ Co-first authors.

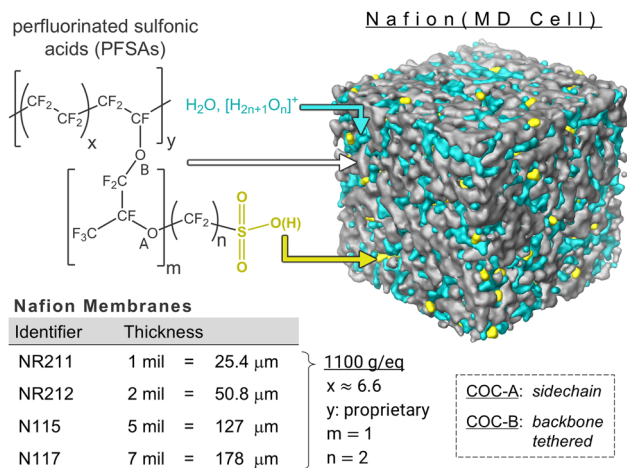


Fig. 1 General perfluorinated sulfonic acid structure (top left) and physical parameters of Nafion membranes (bottom left). Snapshot of hydrated molecular dynamics Nafion cell (top right).

of investigation, a consensus on hydration-dependent size, shape, and interconnectivity of domains remains elusive.^{15,22,26,27}

Exceptional detail of hydration-dependent exchange site environments have been derived from analyses of IR and confocal Raman spectra, correlated to density functional theory (DFT) calculated normal modes and molecular dynamics (MD) simulations (Fig. 2a and b).^{8,28–35} Eigenvector normal mode animations^{36,37} guide vibrational group mode assignments, involving major contributions by multiple functional groups with intimately (mechanically) coupled^{38–40} internal coordinates.^{8,28–35} For example, the PFSA fingerprint region hosts a set of group modes primarily attributed to the mechanically coupled exchange site and its nearest ether link (Table 1).^{8,28–35,37,40–46} These group modes are specific to either the proton associated (*i.e.*, α) or proton dissociated (*i.e.*, β) exchange site states (Fig. 2c). All sites are protonated (*i.e.*, α -sites) in exhaustively dehydrated PFSAs (blue spectra) and contribute to high and low frequency α -bands (*i.e.*, α_{HF} and α_{LF} , respectively). Nearly all sites are deprotonated (*i.e.*, β) in the fully hydrated state (red spectra) and contribute to high and low frequency β -bands (*i.e.*, β_{HF} and β_{LF} , respectively). At partial hydration states (purple spectra), *in situ*,^{8,29,35,47–50} *ex situ*,^{29,30,35,51,52} and *operando*³¹ hydration-dependent spectra show the coexistence of α - and β -bands. A video of the simultaneous evolution of α -bands and the disappearance of β -bands during Nafion dehydration (corresponds to Nafion column of Fig. 2b) is available (ESI Video†). Doubly protonated exchange sites^{19,53–55} (*i.e.*, $-\text{SO}_3\text{H}_2^+$) are assumed not to be statistically significant (Section 3.2.2) and are therefore not considered in this work.

PFSA hydration is often specified by the overall water/ $\text{SO}_3(\text{H})$ ratio, λ .⁵⁶ The exchange site-specific local water content, Λ ,[¶]

represents the number of water molecules within a designated radial boundary ('inner-sphere') surrounding each exchange site.^{8,33} *Ab initio* studies show that exchange site protonation states are Λ -dependent.^{8,29,30,33,35,47,53,57–67} For example, DFT geometry optimizations show that dissociation of an isolated RSO_3H molecule requires at least three waters of solvation (*i.e.*, $\Lambda_{\text{d}} = 3$).^{47,53,60–63,66,67} In contrast, classical MD (CMD) simulations require *a priori* designation of protonation states, as their empirical force fields do not accommodate dynamic proton transfer reactions.^{64,65} Accordingly, Nafion is typically modeled with a minimum λ value of 3,^{68–84} with the assumption that all exchange sites are β for $\lambda \geq 3$.^{68–78,80–88} This approach implies a uniform Λ distribution, contrary to an evolving number of works suggesting non-uniform water content^{8,40,64,79,89–91} and our recent CMD study[¶] showing a wide Λ distribution for all simulated non-zero λ cells.⁸ This approach also presupposes that DFT-optimized Λ environments are sufficiently realistic and applicable to MD-modeled exchange site environments. This paper addresses the validity of this presumption.

Sengupta and Lyulin demonstrated the profound morphological impact of *a priori* designated protonation levels (*i.e.*, α -site/ β -site ratios) in Nafion CMD simulations.⁷⁹ Their work stimulated us to perform reactive force field (ReaxFF)-based MD simulations, which explicitly accommodates modeling of proton dissociation and association processes without requiring *a priori* knowledge of protonation levels.⁹² ReaxFF offers a unique balance between accuracy and computational expense,⁹³ enabling simulations of systems with up to a million atoms for much larger length and time scales than those modeled with empirical valence bond^{20,94–98} and *ab initio* MD simulations.^{21,42,99,100}

In Section 3.1, DFT-calculated normal modes of Nafion repeat units are correlated to acquired spectra by visualization of eigenvector animations. In Section 3.2, ReaxFF MD simulations of Nafion cells with λ values ranging from 0 to 20 are presented. Each hydrated simulation produces equilibrium proportions of α - and β -sites. In Section 3.2.3, stereoelectronic factors that affect exchange site protonation states are discussed. To our knowledge, this is the first report of ReaxFF modeling of proton exchange between ion-exchange sites and water within an ionomer membrane.

The following premises underpin the approach adopted in this work. Given that Λ_{α} and Λ_{β} represent the number of inner-sphere water molecules for protonated (α) and deprotonated (β) sites, respectively:

- **Premise I** – for any overall ionomer water content λ , there are characteristic Λ_{α} and Λ_{β} distributions. The total frequency of each distribution (*i.e.*, the sum of all Λ_{α} and Λ_{β} number frequencies) is λ -dependent.

- **Premise II** – the fingerprint region of any membrane spectrum can be interpreted as the convolution of either:

- (a) The site-spectrum of every exchange site environment (each characterized by some Λ_{α} or Λ_{β}).

§ Total dehydration of ionomers requires a rigorous drying procedure as described by Doan *et al.*²⁹

¶ The symbol, Λ , is read as 'lambda local'. The symbol replaces λ_{site} ³³ and λ_{local} ,⁸ both introduced by Loupe *et al.*

¶ All exchange sites were *a priori* designated β and all $\Lambda \leq \Lambda_{\text{d}}$ sites were assumed to represent α -sites.



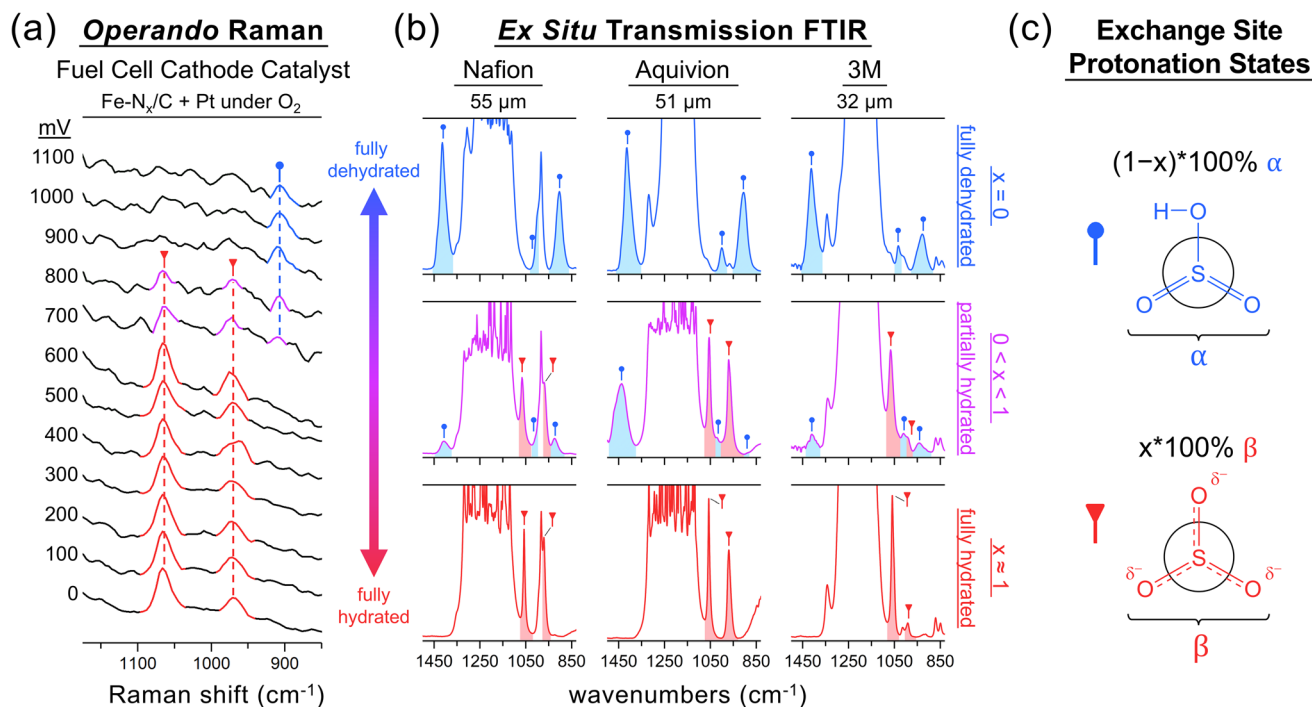


Fig. 2 *Operando* and *ex situ* vibrational spectra. (a) Potential dependent Raman spectra of 25 wt% Nafion cathode catalyst (Fe-N_x/C + Pt under O₂), adapted with permission from Kendrick *et al.*³¹ 2016 under Creative Commons Attribution 4.0 License. (b) Hydration-state dependent PFSA membrane transmission spectra, adapted with permission from Loupe *et al.*⁸ Copyright 2020 American Chemical Society. (c) α- and β-site Newman projections.

(b) Two subspectra:¹⁰¹ an α-subpectrum (convolution of all Λ_α site-spectra) and a β-subpectrum (convolution of all Λ_β site-spectra).

• **Premise III** – the IR spectrum of an ionomer membrane is characteristic of its overall water content λ.

The Fig. 2b spectra exemplify the premises: Any dry membrane spectrum (blue) has no β-subpectrum component; any fully hydrated membrane spectrum (red) has a negligible α-subpectrum component; any partially hydrated spectrum (purple) is a convolution of an α-subpectrum and β-subpectrum.

2 Methods

2.1 DFT calculations

Unrestricted hybrid DFT (PBE0-D3/LACV3P**++) was used for all quantum mechanics (QM) calculations including geometry optimization and IR vibrational normal mode analysis (*i.e.*, frequency analysis). Calculations were performed in Jaguar (v. 12.1, ©Schrödinger, LLC, New York, NY, 2024) and executed on a high-performance computing cluster (Discovery cluster, Holyoke, MA) accessed through Northeastern University. The D3 semi-empirical correction¹⁰² was paired with the PBE0

Table 1 Vibrational group mode assignments for Nafion 212 transmission FTIR bands and corresponding DFT-calculated normal modes (X3LYP/6-311G**++), adapted with permission from Loupe *et al.*⁸ copyright 2020 American Chemical Society. Eigenvector animations for each normal mode are available^{8a}

Frequency (cm ⁻¹)			
FTIR	DFT	Label	Group mode assignment (dominant contributors in bold)
1414	1405	α _{HF}	ν _{as} (SO₃H) + ν _{as} (COC-A)
1058	1059	β _{HF}	ν _{as} (COC-A) + ν _s (SO₃⁻)
969	983	β _{LF}	ν _s (SO₃⁻) + ν _{as} (COC-A)
910	786	α _{LF}	ν _s (SO₃H) + ν _s (COC-A)

^a ν_s: symmetric stretching; ν_{as}: asymmetric stretching; HF: high-frequency; LF: low-frequency.



hybrid functional^{103–105} to account for long-range van der Waals attraction interactions. Polarization (*i.e.*, “**”) ¹⁰⁶ and diffuse (*i.e.*, “++”) ¹⁰⁷ functions were added to the triple- ζ Gaussian-type basis set LACV3P.**

Three Nafion ‘component models’ (perfluorinated dimethyl ether, triflic acid, and triflate) and three Nafion monomer models (1-*mer*-sbb, 1-*mer*-CH₃, and 1-*mer*-CF₃) were the model compounds¹⁰¹ considered (Fig. S1-1; ESI†). The component models represent minimal ionomer fragments, making them useful for elucidating individual functional group contributions to ionomer vibrational normal modes (ESI-S1†).⁸ All models were optimized without an external dielectric medium or implicit solvent.^{30,35} QM calculations were performed on incrementally hydrated 1-*mer*-CF₃ as previously reported for triflic acid.^{8,30,35}

2.2 MD simulations

2.2.1 System setup and equilibration. Nafion membranes were modeled using 32 10-*mer* chains (*i.e.*, 320 total exchange sites per simulation cell) at nine hydration levels: $\lambda = 0, 1, 2, 3, 5, 7, 10, 15, 20$ (Fig. 3). Each monomer has 14 –CF₂– units followed by a single sidechain and an additional CF₂ unit. Oligomer backbones are capped with fluorine atoms. All exchange sites are protonated (and remain so) during the non-reactive equilibration stage. The above results in 692 atoms per oligomer and 22144 Nafion atoms per cell. The total number of atoms per cell varies from 22144 ($\lambda = 0$) to 41344 ($\lambda = 20$). Initial cell structures were generated using Packmol.¹⁰⁸

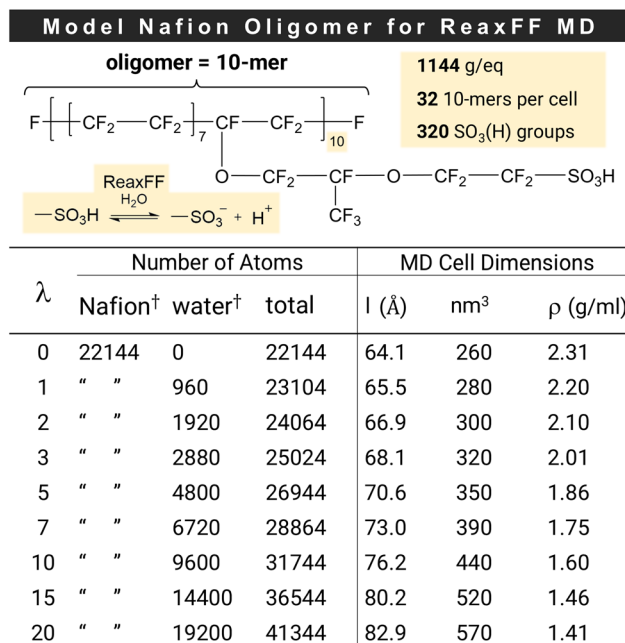
All cells were first relaxed by steepest-descent energy minimization, followed by a 10 ps NVT^{17,109} simulation at 10 K to generate initial velocities. All cell densities were then set to an initial 1.8 g cm^{–3} (N117 at $\lambda = 15$ is ~ 1.75 g cm^{–3})¹⁶ before undergoing the following 1.6 ns annealing procedure: **1a**) heating from 10 K to 400 K over 150 ps; **1b**) 100 ps equilibration at 400 K; **2a**) cooling to 300 K over 50 ps; **2b**) 100 ps equilibration; **2c**) heating to 400 K over 50 ps; **2d**) 100 ps equilibration; **2e**) three repetitions of steps **2a–2d**; **3a**) cooling to 300 K over 50 ps; **3b**) 100 ps equilibration. The cells were then equilibrated for another 10 ns at 300 K.

The universal force field¹¹⁰ was employed for the non-reactive simulations using LAMMPS.^{111,112} The equations of motion were integrated using the Verlet algorithm¹¹³ (1.0 fs time step). Electrostatic interactions were calculated with the particle-particle mesh method.¹¹⁴

2.2.2 ReaxFF MD simulations. ReaxFF, developed by Goddard and co-workers, describes bond formation and breaking in terms of a bond energy and bond order *versus* distance relation, which allows bond order changes within MD simulations.⁹² ReaxFF partitions the total system energy (E_{system}) into bonded and nonbonded terms (eqn (1)).

$$E_{\text{system}} = E_{\text{bonded}} + E_{\text{nonbonded}} \quad (1)$$

Bonded terms include bond, over-coordination penalty (over), under-coordination stability (under), lone-pair (lp),



[†] Initial number of atoms before initiation of ReaxFF simulation.

Fig. 3 Nafion oligomer for molecular dynamics simulations (top) and table of specified number of atoms, cell dimensions, and densities (bottom).

valence angle (val), and torsion (tor) energies. Nonbonded terms include Coulombic (Coul), van der Waals (vdW), and hydrogen bond (HB) energies (eqn (2)).

$$E_{\text{system}} = (E_{\text{bond}} + E_{\text{over}} + E_{\text{under}} + E_{\text{lp}} + E_{\text{val}} + E_{\text{tor}}) + (E_{\text{Coul}} + E_{\text{vdW}} + E_{\text{HB}}) \quad (2)$$

ReaxFF determines bonded interactions among all atoms in the system using the bond order (BO) concept, where BOs are expressed as a continuous function of interatomic distance. Contributions from σ , π , and $\pi\pi$ bonds are calculated as follows:

$$\begin{aligned} \text{BO}_{ij} &= \text{BO}_{ij}^{\sigma} + \text{BO}_{ij}^{\pi} + \text{BO}_{ij}^{\pi\pi} \\ &= \exp \left[p_{\text{bo}1} \left(\frac{r_{ij}}{r_0^{\sigma}} \right)^{p_{\text{bo}2}} \right] + \exp \left[p_{\text{bo}3} \left(\frac{r_{ij}}{r_0^{\pi}} \right)^{p_{\text{bo}4}} \right] \\ &\quad + \exp \left[p_{\text{bo}5} \left(\frac{r_{ij}}{r_0^{\pi\pi}} \right)^{p_{\text{bo}6}} \right] \end{aligned} \quad (3)$$

where BO_{ij}^{σ} , BO_{ij}^{π} , and $\text{BO}_{ij}^{\pi\pi}$ are the partial contributions of σ , π , and $\pi\pi$ bonds, respectively, between atoms *i* and *j* (r_{ij} is the distance between *i* and *j*). The σ , π , and $\pi\pi$ bond radii are r_0^{σ} , r_0^{π} , and $r_0^{\pi\pi}$, respectively. The p_{bo} terms are empirical parameters fitted against reference QM results (or sometimes Experimental data). The nonbonded terms (*i.e.*, Coul, vdW, and HB) are calculated between every pair of atoms regardless of their connectivity. The QEq charge equilibration method¹¹⁵ was used for electrostatic interactions. A more comprehensive review of ReaxFF is described elsewhere.⁹³

If all charges are allowed to respond to the forces and electric fields, the proper dielectric constant is 1. Accordingly, our DFT calculations of model compounds employ a dielectric constant

** LACV3P is identical to 6-311G for atoms lighter than K.



of 1 (Section 2.1). Typical force fields use fixed point charges and hence, should have a non-unit dielectric constant, but the correct value is generally unknown. UFF and ReaxFF allow charges to fluctuate according to the QEq charge equilibration model. Thus, the proper dielectric constant is 1, just as for QM. Applying macroscopic dielectric constants to atomistically modeled environments is not justified.

In this study, ReaxFF was trained to capture the correct qualitative trends in reactivity and proton dynamics in hydrated polymeric systems, based on QM data for triflic acid in the presence of one to five explicit water molecules. ReaxFF-minimized structures show that proton transfer occurs when three or more water molecules are present, in good agreement with QM results (Fig. S2-1; ESI†).^{47,53,60–63,66,67}

ReaxFF MD (RMD) simulations were conducted on all classically equilibrated cells (Section 2.2.1) to simulate proton exchange between Nafion and water. The cells were relaxed by steepest-descent energy minimization, followed by a 10 ps NVT simulation at 10 K. The cells were then heated to 300 K over 50 ps. ReaxFF simulations were conducted for 2 ns under NPT¹⁷ conditions (300 K and 1 atm), during which the cell densities were monitored (Fig. S2-2; ESI†).

RMD simulations were performed using LAMMPS (0.5 fs time step). RMD structure files corresponding to snapshots at 2 ns are available (ESI PDB files†). Molecular surfaces were generated in Maestro (version 13.3, ©Schrödinger, LLC, New York, NY, 2023). Closed Connolly surfaces with a 1.4 Å probe radius were applied to various atom groupings.

3 Results and discussion

3.1 DFT calculated Nafion monomer vibrational normal modes

We have reported extensive frequency analyses (at the X3LYP/6-311G**++ level) of dry (*i.e.*, $\Lambda = 0$) RSO₃(H) models (*e.g.*, 1-*mer*-CH₃; Fig. 4a).^{8,29–31,33–35} These include group mode assignments (Table 1) correlated to hydration-dependent PFSA vibrational bands (Fig. 2a and b). Assignments are based on the visualization of eigenvector normal mode animations (ESI Videos†).^{8,30,36,37,40} Excellent agreement was found between X3LYP 1-*mer*-CH₃ frequencies and corresponding IR†† band frequencies (Fig. 4b). The calculated frequencies are within ~1% of experimental values, except for α_{LF} , which is 14% lower than the observed 910 cm⁻¹ band. Frequency errors are often attributed to scaling factors (varies with functional and basis set) and the choice of chemical repeat unit.^{28,35,101}

Yamaguchi and Ohira evaluated several hybrid functional and basis set combinations for PFSA sidechain models, finding that PBE0/6-311G(d) offers the best balance of accuracy and computational time.³⁷ Thus, we performed frequency analyses with the PBE0-D3 hybrid functional,^{103–105} which accounts for long-range interactions,¹⁰² and the same basis set as previously reported^{8,29–31,33–35} (Section 2.1). The 1-*mer*-CH₃ α_{LF} value improves by ~6% and the β_{LF} value improves slightly with PBE0-D3 (Fig. 4b).

All other listed modes increase in error. The insensitivity of the Nafion side chain (NSC)†† band to state-of-hydration^{8,30,35} (band is fixed at 983 cm⁻¹) is better captured with PBE0-D3. Yet, the X3LYP NSC values are much closer to the observed band. Thus, for 1-*mer*-CH₃, neither functional clearly outperforms the other, deeming both suitable for frequency analysis.

Historically, the 1-*mer*-CH₃ model was used to avoid “computational interference” with the sidechain –CF₃ group.^{8,29,33–35} Replacing methyl groups with –CF₃ groups (1-*mer*-CF₃; Fig. 4a), however, has a negligible effect on the PBE0-D3 frequencies (Fig. 4b). Notably, the NSC group modes (dominated by sidechain –CF₃ and COC–B motions) are unaffected by the backbone capping groups, as their large separation from the COC–B group prevents significant coupling of their motions.

The exceptional accuracy of the X3LYP-calculated dry 1-*mer*-CH₃ β_{HF} frequency (1059 cm⁻¹; Fig. 4b) is fortuitous given that β -sites are formed and stabilized in the presence of water.^{47,53} In comparison, the PBE0-D3 β_{HF} frequency for all monomer models are 26 cm⁻¹ less than the observed 1058 cm⁻¹ band. Hydration of the 1-*mer*-CF₃ β -site greatly reduces the PBE0-D3 β_{HF} frequency error ((Table S1-2†): β_{HF} increases to 1049 cm⁻¹ at Λ_{d} (*i.e.*, $\Lambda = 3$); β_{HF} is 1052 ± 1.5 cm⁻¹ between $\Lambda = 3$ and $\Lambda = 15$. As Λ increases, the local permittivity around the β -site should approach that of hydrated membrane β -sites,^{10,116} allowing for a more realistic β_{HF} group mode calculation.

DFT-calculated exchange site environments are typically derived from a limited landscape of energy-minimized structures,⁵⁷ lacking both the structural complexity and site-to-site variation inherent in real ionomer repeat units.¹⁰¹ Thousands of realistic exchange site environments are accessed with a lower level of theory like MD, as discussed below (Section 3.2).

QM-calculated IR spectra of all model compounds (ESI-S1†) and normal mode animations for component models and 1-*mer*-CF₃ are available (ESI Videos†).

3.2 Nafion ReaxFF MD simulations

RMD simulations of nine Nafion cells were performed at λ values ranging from 0 to 20. Trajectories of protonation levels *versus* time established a standard simulation duration of 2 ns (Fig. S2-3; ESI†). In this section, ‘water’ includes all water molecules whether they are neutral or are part of a protonated hydration complex (*i.e.*, hydronium, Zundel, Eigen, *etc.*),^{21,73} unless otherwise specified. A standard sulfur-centered radial cutoff distance (*b*) of 6 Å§§ defines an ‘inner-sphere’ and ‘outer-sphere’ boundary in this work. The integer number of inner-sphere waters is Λ . Each DFT and RMD-generated exchange site environment will be referred to by its protonation state (*i.e.*, α or β) and a numerical subscript representing the site’s Λ value (*e.g.*, β_{14} denotes an ionized exchange site with 14 inner-sphere waters).

3.2.1 Inner-sphere and outer-sphere waters. For real and theoretical ionomer samples, W_{tot} is the total number of waters (eqn (4)):

†† NSC group modes are dominated by $\delta_{\text{u}}(\text{CF}_3)$ (see ESI Videos†).

§§ The arbitrarily chosen 6 Å cutoff boundary contains all 1-*mer*-CF₃ hydration waters (Fig. S1-6; ESI†).

†† Transmission FTIR of Nafion 212 (NR212; see Fig. 1 for membrane specifications).



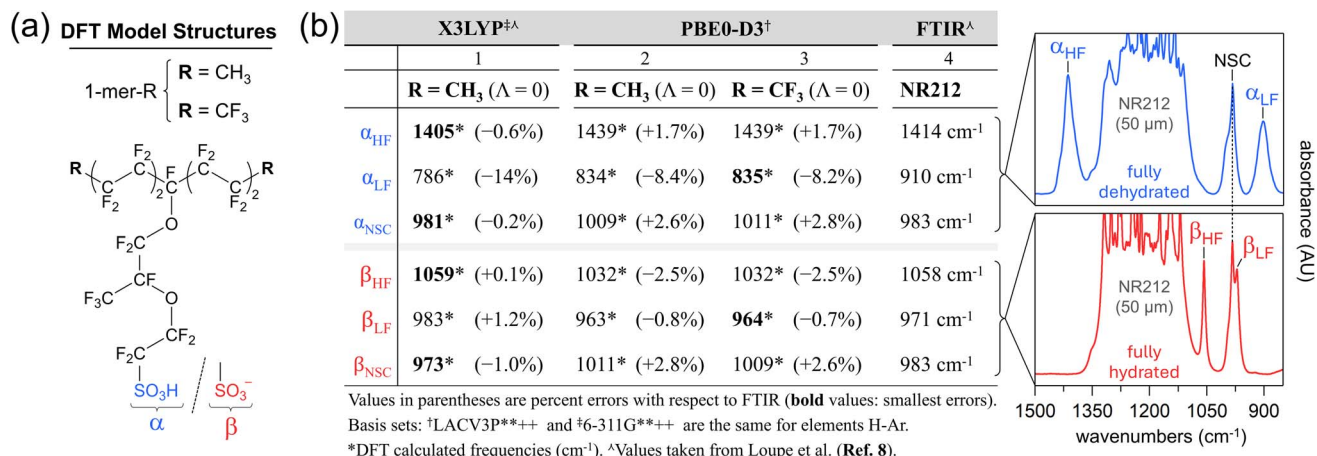


Fig. 4 Theoretical and experimental Nafion vibrational group mode analyses: (a) model monomer structures; (b) effect of functional and backbone capping groups on calculated frequencies (columns 1–3). Animations for all listed 1-mer-CF₃ normal modes (column 3) are available (ESI[†]). X3LYP-based frequencies (column 3), FTIR band frequencies (column 4), and NR212 spectra are adapted with permission from Loupe et al.⁸ Copyright 2020 American Chemical Society.

$$W_{\text{tot}} = \lambda N_A \frac{m_{\text{dry}}}{EW} = \lambda N_S \quad (4)$$

$$n_{\text{over}} = \sum_{j=2}^q f_j(j-1) \quad (8)$$

where N_A is Avogadro's number, m_{dry} is the dry mass, and N_S is the total number of sulfur atoms. For both the RMD simulations and our previous all-classical simulations, N_S is 320.⁸ Every exchange site has an integer Λ value representing inner-sphere water oxygens. Waters contributing to multiple Λ (*i.e.*, multiple sulfur-spheres) are “bridge”⁸⁴ waters. Waters that contribute to single sulfur-spheres are “non-bridge” waters. The sum of all inner-sphere waters (W_{IS}) and all outer-sphere waters (W_{OS}) is W_{tot} (eqn (5)).

$$W_{\text{tot}} = W_{\text{IS}} + W_{\text{OS}} \quad (5)$$

Fig. 5a shows histograms of Λ number frequencies (f_{Λ}) versus Λ for the nine simulated cells, where f_{Λ} values range from 0 to N_S . Thus, the range of f_{Λ} values can formally be defined by the set, $\{f_{\Lambda} \in \mathbb{N}_0 | f_{\Lambda} \leq N_S\}$. The histogram color coding is discussed later (Section 3.2.2). The sum of the f_{Λ} by Λ products is the MD cell's histogram number, n_{hist} (eqn (6)).

$$n_{\text{hist}} = \sum_{\Lambda=0}^p f_{\Lambda} \Lambda \quad (6)$$

The upper limit p is the maximum Λ value recorded for a Λ distribution, which depends on b and λ (*e.g.*, p is 19 for $\lambda = 20$; Fig. 5a). The domain of Λ is formally, $\{\Lambda \in \mathbb{N}_0 | \Lambda \leq p\}$. If there are no bridge waters, n_{hist} equals W_{IS} . The presence of bridge waters breaks the equality and thus contributes towards a net overcount (n_{over}) of W_{IS} (eqn (7)).

$$W_{\text{IS}} = n_{\text{hist}} - n_{\text{over}} \quad (7)$$

Bridge-waters within j inner-spheres are overcounted $j - 1$ time(s), where j is 2 or larger. The domain of j is formally, $\{j \in \mathbb{N} | j \geq 2\}$. The number of bridge waters occupying j spheres is f_j . The sum of f_j by $j - 1$ products is n_{over} (eqn (8)).

The upper limit q is the maximum number of inner-spheres that a water resides in, which, similar to p , depends on b and λ (q is either 4 and 5 across the nine λ systems, Table S2-1; ESI[†]). A more descriptive W_{IS} definition is now obtained by substitution of eqn (6) and (8) into eqn (7) (eqn (9)).

$$W_{\text{IS}} = \sum_{\Lambda=0}^p f_{\Lambda} \Lambda - \sum_{j=2}^q f_j(j-1) \quad (9)$$

A full expression for W_{tot} is established by substitution of eqn (9) into eqn (5) (eqn (10)).

$$W_{\text{tot}} = \sum_{\Lambda=0}^p f_{\Lambda} \Lambda - \sum_{j=2}^q f_j(j-1) + W_{\text{OS}} \quad (10)$$

The water-count discrepancy (δ) represents the amount by which n_{hist} is an overcount ($\delta > 0$) or undercount ($\delta < 0$) of W_{tot} , relative to W_{tot} (eqn (11)).

$$\delta = \frac{n_{\text{hist}} - W_{\text{tot}}}{W_{\text{tot}}} \times 100\% \quad (11)$$

Alternatively, δ may be recast in terms of n_{over} and W_{OS} (total number of over- and under-counts, respectively) by substitution of eqn (5) and (7) into eqn (11) (eqn (12)).

$$\delta = \frac{n_{\text{over}} - W_{\text{OS}}}{W_{\text{tot}}} \times 100\% \quad (12)$$

Eqn (4)–(12) are summarized as a two-dimensional conceptual illustration (Fig. S2-4; ESI[†]).

Fig. 5b shows δ versus λ for the simulations of the eight hydrated systems. There are two linear segments with slopes of



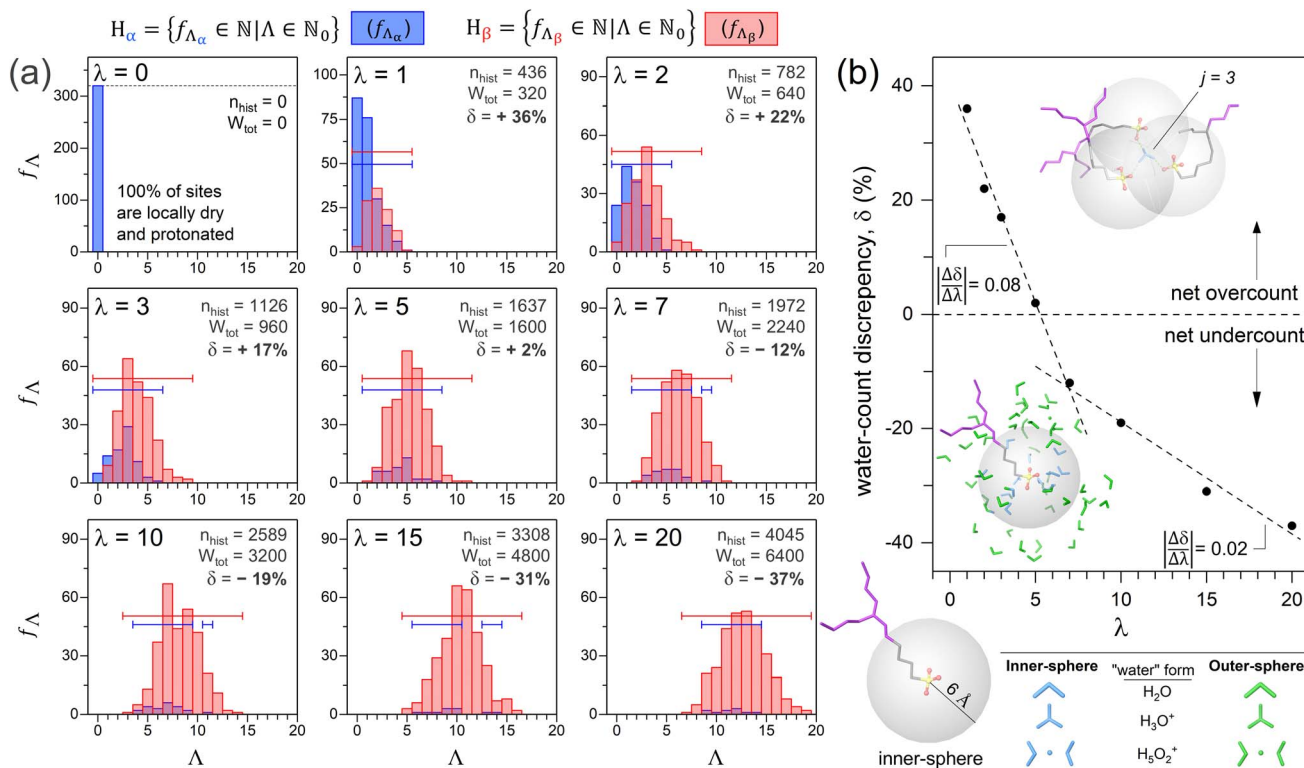


Fig. 5 Inner- and outer-sphere water content versus λ . (a) Overlapping histograms of f_{Λ} versus Λ , for nine λ values. Blue and red histograms correspond to protonated and deprotonated exchange sites, respectively. (b) Water-count discrepancy, δ , versus λ . Structural examples contributing to δ are inset: (top) protonated bridge-water overcounted twice (*i.e.*, $j = 3$); and (bottom) non-bridge inner-sphere waters (light blue) and undercounted outer-sphere waters (green).

−0.08 (between $\lambda = 1$ and $\lambda = 7$) and −0.02 (between $\lambda = 7$ and $\lambda = 20$) that intersect at λ_7 . The steeper segment results from a reduction of overlapping sulfur-spheres (*i.e.*, n_{over} decreases), concurrent with the emergence of a water-rich phase (*i.e.*, W_{OS} increases). For very low λ , waters are predominantly inner-sphere¹⁰ and bridge exchange sites. Fig. 6 depicts bridge waters as light blue spheroids bridging yellow trefoils (exchange sites). Protonated bridge waters, or “hydronium-mediated bridges”, have been reported in CMD simulations of Nafion, Aquivion, and a perfluorinated phosphoric acid by Jang and coworkers.^{69,83} The $\lambda = 0$ and $\lambda = 2$ cell cross-sectional surfaces both show a greater extent of clustered exchange sites than that of $\lambda = 20$, consistent with reports of stronger S–S pairwise correlations with decreasing λ .^{69,71,76–78,83,84,87} At $\lambda = 5$, while n_{over} is counterbalanced by W_{OS} , there remain bridge waters that impact the slope up until the intersection at $\lambda = 7$. The shallower slope is attributed almost exclusively to the progressive increase of outer sphere waters (*i.e.*, W_{OS} increases while n_{over} is negligible). Outer-sphere waters (green) are either bulk-like (*i.e.*, clustered) or isolated (Fig. 6). The hydration-dependent growth of a three-phase nanomorphology is illustrated in Fig. S2-5 (ESI†).

3.2.2 Λ_{α} and Λ_{β} distributions. The $\lambda = 0$ cell (*i.e.*, $W_{\text{tot}} = 0$) has 320 locally dry exchange sites (*i.e.*, $f_0 = 320$), corresponding to the single $\Lambda = 0$ bar in Fig. 5a. All sites remained protonated throughout the simulation period (Fig. S2-3; ESI†), consistent

with no autoprotolysis products (*i.e.*, $-\text{SO}_3\text{H}_2^+$ and $-\text{SO}_3^-$).^{19,53–55} A solvent autoprotolysis constant (K_{ap}), depends on the solvent equivalent weight, density, and dielectric constant (κ).¹¹⁷ Although K_{ap} for Nafion is not reported, it is likely negligible because of its low κ ($\kappa \approx 4$ for $\lambda = 1$),¹¹⁶ high equivalent weight, and low anhydrous ionic conductivity.^{118,119} Even for sulfuric acid, which has a high κ ($\kappa \approx 100$; $K_{\text{ap}} = 2.4 \times 10^{-4}$ at 25 °C)^{120,121} and low equivalent weight, only three H_3O_4^+ ions are expected at equilibrium in a $\lambda = 0$ cell volume (ESI-S3†). In a transmission FTIR study of ‘dry’ Nafion, Buzzoni *et al.* attributed a weak intensity 1060 cm^{-1} band (β_{HF}) to $-\text{SO}_3^-$ groups formed *via* autoprotolysis.¹²² This is unlikely given the above discussion of sulfuric acid. In an ATR-FTIR study of ‘dry’ Nafion by Negro *et al.*, the 1060 cm^{-1} band is attributed to $-\text{SO}_3^-$ groups persisting due to residual membrane water.¹²³ Whatever the acquisition technique, complete absence of the 1060 cm^{-1} band requires maintenance of a completely dry atmosphere during measurement.²⁹ Loupe *et al.* describes in detail the challenges of obtaining completely dry PFSA spectra in an ATR configuration.⁸

All non-zero λ cells result in α -site and β -site Λ distributions, visualized as blue and red histograms, respectively (Fig. 5a). Each histogram is represented as a numerical set, with each element corresponding to a single bar. For example, each blue histogram is represented by the set H_{α} (eqn (13)).

$$H_{\alpha} = \{f_{\Lambda_{\alpha}} \in \mathbb{N} | \Lambda \in \mathbb{N}_0\} \quad (13)$$



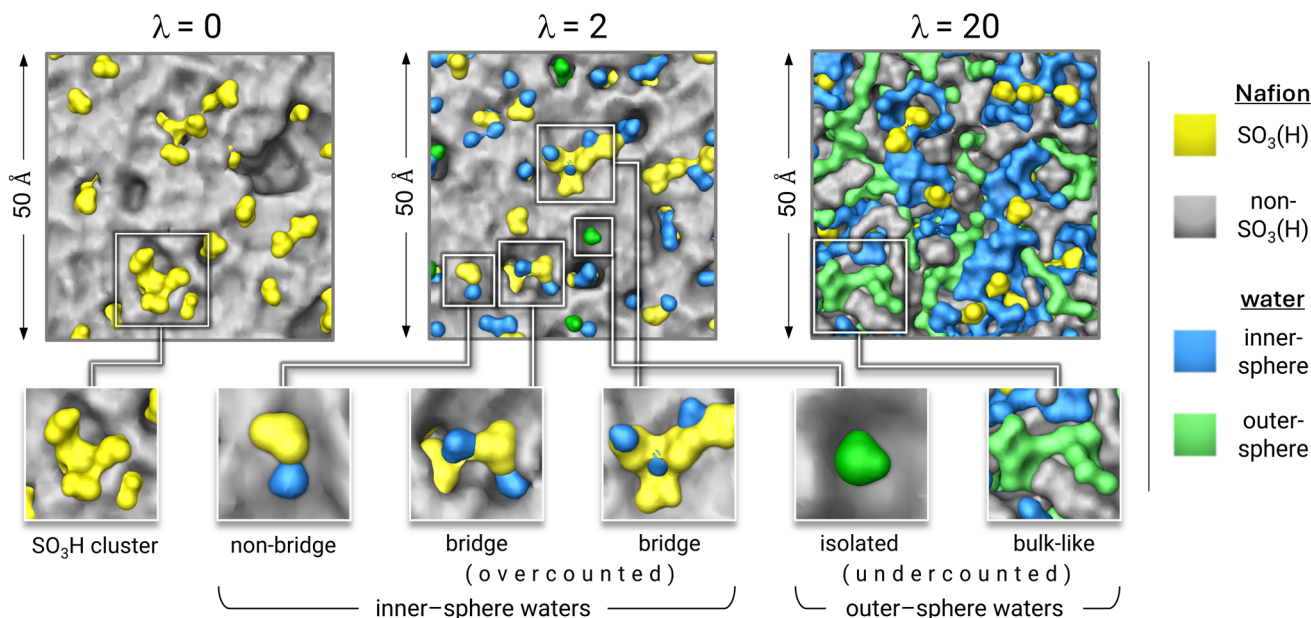


Fig. 6 Colored Connolly surfaces applied to atom groupings from $\lambda = 0, 2$, and 20 MD cells. Inner-sphere waters (light blue) are either bridge (overcounted) or non-bridge. Outer-sphere waters (green) are either isolated or bulk-like (both are undercounted).

Each H_α element corresponds to a blue bar, representing non-zero frequencies of α -sites with specific Λ values (f_{Λ_α}). The number of f_{Λ_α} elements (*i.e.*, number of blue bars) is the cardinality of H_α , $|H_\alpha|$ (Table 2). The sum of f_{Λ_α} over all Λ represents the total number of α -sites in the cell (N_{S_α} ; eqn (14)).

$$N_{S_\alpha} = \sum_{\Lambda=0}^p f_{\Lambda_\alpha} \quad (14)$$

Similarly, each red histogram is represented as the set, H_β (eqn (15)).

$$H_\beta = \{f_{\Lambda_\beta} \in \mathbb{N} | \Lambda \in \mathbb{N}_0\} \quad (15)$$

Each H_β element corresponds to a red bar, representing non-zero frequencies of β -sites with specific Λ values (f_{Λ_β}). The number of f_{Λ_β} elements is the cardinality of H_β , $|H_\beta|$ (Table 2). The sum of f_{Λ_β} over all Λ is the total number of β -sites for the cell (N_{S_β} , eqn (16)).

$$N_{S_\beta} = \sum_{\Lambda=0}^p f_{\Lambda_\beta} \quad (16)$$

The sum of N_{S_α} and N_{S_β} is N_S (eqn (17) and (18)).

$$N_S = N_{S_\alpha} + N_{S_\beta} = \sum_{\Lambda=0}^p (f_{\Lambda_\alpha} + f_{\Lambda_\beta}) \quad (17)$$

The sum of corresponding f_{Λ_α} and f_{Λ_β} values is f_Λ , which represents the number of exchange sites with a given Λ , independent of protonation state. The rightmost expression of eqn (17) is then simplified to represent the sum of every Fig. 5a histogram bar, independent of its color (eqn (18)).

$$N_S = \sum_{\Lambda=0}^p f_\Lambda \quad (18)$$

Fig. 7 shows N_{S_α} versus λ . According to eqn (14), the value of each bar is the sum of all blue histogram bars from Fig. 5a, for the corresponding λ . Thus, the exponential decay of N_{S_α} versus λ represents the diminishing significance of blue histograms with increasing water content. The persistence of α -sites across all λ contradicts the widely adopted notion that all exchange sites are ionized for $\lambda \geq 3$.^{68–78,80–87} Our finding concurs with Sengupta and Lyulin's report of α -sites at high λ .⁷⁹

All eight hydrated cells result in overlapping histograms across variable Λ intervals (Fig. 5a). For λ_1 , both H_α and H_β span Λ values from 0 to 5 (*i.e.*, $|H_\alpha| = |H_\beta| = 6$). With increasing λ , both histograms shift towards higher Λ , signifying a growing number of inner-sphere waters for every exchange site. Consequently, the proportion of locally dry sites (*i.e.*, f_0/N_S) diminishes from 100% to 2% between $\lambda = 0$ and $\lambda = 3$. For $\lambda = 5$ and beyond, all exchange sites are hydrated by at least one water. It is noteworthy that the absence of $\Lambda = 0$ sites starting at $\lambda = 5$, coincides with the equality of over- and undercounted waters (*i.e.*, $\delta \approx 0$ at $\lambda = 5$, Fig. 5b).

Table 2 shows that $|H_\beta|$ generally increases with λ , reflecting a growing variety of β -site Λ values with hydration. This is consistent with a 3-mer Aquivion model, which shows an increasing preference for non-uniform hydration as λ increases.⁸⁹ In contrast, $|H_\alpha|$ remains roughly the same across all non-zero λ . The number of Λ values over which H_α and H_β overlap also show little variation with λ (Fig. 5a). The tails of the α histograms, on the high Λ side, never overlap with the β histograms beyond $\lambda = 1$, indicating that sites with the largest Λ values in any cell are always ionized.



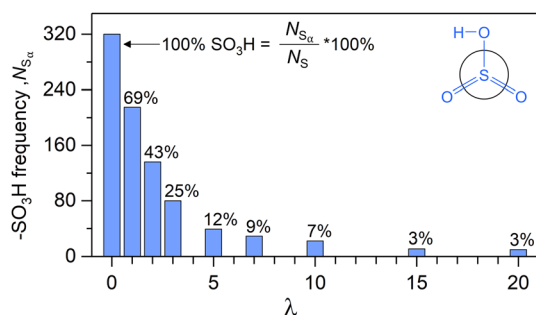
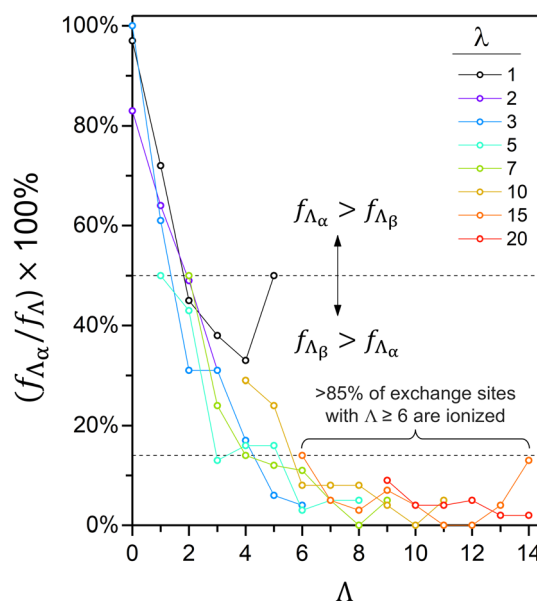
Table 2 Cardinality of H_α , H_β , and the union of both sets versus λ for the nine simulated Nafion cells

λ	0	1	2	3	5	7	10	15	20
$ H_\alpha $	1	6	6	7	8	7	7	7	6
$ H_\beta $	0	6	9	10	11	10	12	12	13
$ H_\alpha \cup H_\beta $	1	12	15	17	19	17	19	19	19

For any non-zero λ , the proportion of α -sites with a given Λ generally decreases with increasing Λ (*i.e.*, $f_{\Lambda_\alpha}/f_\Lambda$; Fig. 8). In the $\lambda = 3$ cell, for example, $f_{\Lambda_\alpha}/f_\Lambda$ decreases from 61% to 4% as Λ increases from 1 to 6. In every hydrated cell, $f_{\Lambda_\alpha}/f_\Lambda$ is generally less than 15% if Λ is at least 6. The negative correlation between $f_{\Lambda_\alpha}/f_\Lambda$ and Λ is consistent with reports of an increased ionization propensity for $\text{CF}_3\text{SO}_3\text{H}$ (*i.e.*, triflic acid) and $\text{CF}_3\text{CF}_2\text{SO}_3\text{H}$ as Λ increases.^{47,53}

3.2.3 ReaxFF MD exchange site environments. Our RMD simulations model the spatial and temporal heterogeneity of 320 exchange sites per λ (Section 3.2.1 and 3.2.2), where each site is embedded in a realistic environment with up to $\sim 40\,000$ surrounding atoms. The coexistence of α - and β -sites with identical Λ values in all hydrated simulation cells (Fig. 8) suggests that additional variables beyond Λ influence protonation states. The interplay of three stereoelectronic variables is identified through visualization of inner- and outer-sphere environments (Fig. 9): (i) the local number density of polar and non-polar fragments, (ii) the spatial arrangement of water, polymer fragments, and ions, and (iii) long-range electrostatic effects.

The ionization propensity (*i.e.*, acidity) of exchange sites is primarily influenced by the identity and number density of inner-sphere atoms (variable i). Exchange site acidity is expected to increase with Λ (water molecules are highly polar) and decrease with the number density of non-polar fragments within the inner-sphere (*e.g.*, CF_2 and COC groups). We chose to enumerate the latter parameter by the number of non- $\text{SO}_3(\text{H})$ Nafion atoms (N_{np}). Consider, the sixteen 1-*mer*- CF_3 monomer models geometry-optimized with DFT, each with a unique Λ ranging from 0 to 15 (Fig. S1-6; ESI†). Since the monomer N_{np} values vary minimally (Fig. S1-7b; ESI†), their acidities primarily depend on Λ , leading to the establishment of a Λ dissociation threshold ($\Lambda_d = 3$): the exchange site is α for $\Lambda < 3$ and β for $\Lambda \geq 3$

Fig. 7 $-\text{SO}_3\text{H}$ counts versus λ after 2 ns of ReaxFF.Fig. 8 Proportion of protonated (*i.e.*, α) exchange sites versus Λ for all $\lambda > 0$ cells.

(Fig. S1-6; ESI†). In contrast, the RMD-modeled inner-spheres exhibit larger and more diverse N_{np} values that vary independently of Λ , precluding the establishment of a Λ_d (Fig. 9). The RMD inner-sphere structural diversity is facilitated by the abundant presence of polymer fragments from neighboring monomers, with minimal disruption to native³⁴ backbone dihedral angles.^{17,37,57,58,124}

The spatial arrangement of water, polymer atoms, and ions (variable ii) is also expected to influence exchange site protonation states. Consider the two arbitrarily selected α_5 and β_5 sites from the $\lambda = 5$ cell (Fig. 9). The α_5 site proton is partially stabilized by its nearest ether linkage, rather than by a water molecule. Moreover, the α_5 site is directly hydrated by only a single water, while the four remaining waters contribute to the stabilization of a neighboring β -site and two other protons. In contrast, the β_5 site has a much smaller N_{np} , lacks excess charge carriers, and is directly hydrated by two waters. The $\lambda = 2$ cell hosts a nearly equal number of α_2 and β_2 sites (Fig. 8). Despite the selected β_2 site (Fig. 9) having a large N_{np} , two nearby exchange sites contribute to stabilizing the dissociated proton (the proton serves as a hydronium-mediated bridge^{69,83} between three exchange sites). Similarly, Paddison and Elliot report that the number of waters required to cause exchange site ionization is less than three when exchange sites are nearby one



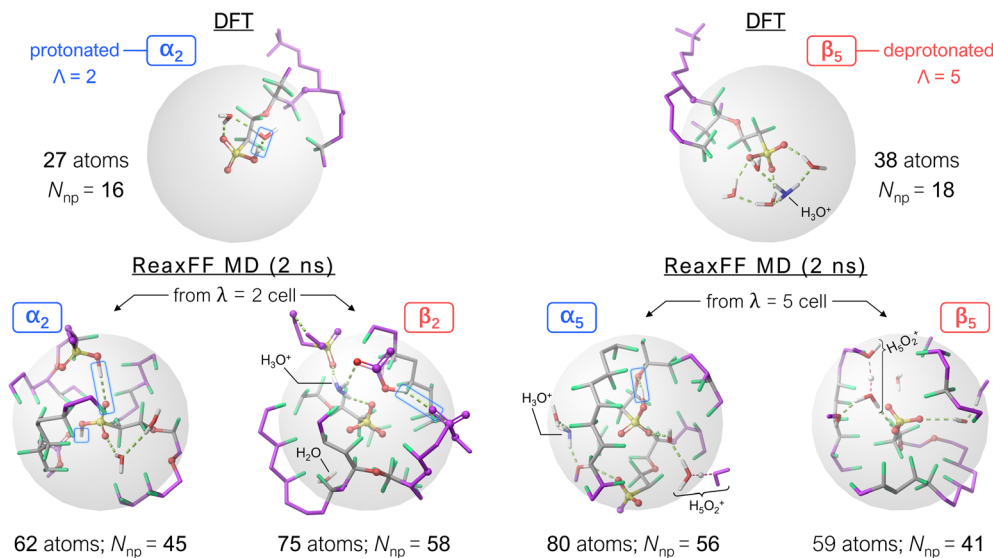


Fig. 9 DFT and ReaxFF MD generated exchange site inner-sphere environments. The number of non-SO₃H inner-sphere atoms is N_{np} (np stands for nonpolar). Not all outer-sphere atoms (purple) are shown. Atom colors: hydrogen (white), carbon (grey), non-hydronium oxygen (red), hydronium oxygen (blue), fluorine (green), sulfur (yellow). An animated version of this figure is available (ESI†).

another.^{57,58} The inclusion of a high dielectric constant continuum solvation model has also been shown to decrease the number of waters required for ionization.⁶¹ Meanwhile, the selected α_2 site proton lacks any direct interaction with water, causing it to remain associated. These observations from arbitrarily selected exchange sites highlight how the highly variable spatial arrangement of inner-sphere species influence proton association and dissociation.

Finally, the exertion of long-range electrostatic forces (variable **iii**) on RMD sites should not be excluded as a variable that influences protonation states. While these forces may be individually insignificant compared to shorter-range Coulombic interactions, the MD cells contain significantly more atoms contributing to them than the 1-*mer*-CF₃ complexes. The overall water content (*i.e.*, λ) is expected to impact the long-range forces exerted on any exchange site.

4 Conclusion

This is the first report on ReaxFF modeling of proton exchange between water molecules and ion-exchange sites within an MD framework. MD simulations were performed on the per-fluorinated sulfonic acid (PFSA) ionomer Nafion at λ values ranging from 0 to 20. Over each 2 ns simulation, proton exchange between initially protonated exchange sites (sulfonic acid groups) and water molecules resulted in λ -dependent equilibrium proportions of protonated sites (α , *i.e.*, sulfonic acid) and deprotonated sites (β , *i.e.*, sulfonate ion). An exponential decrease in the proportion of α -sites was observed with increasing λ . Notably, we found that α -sites persist across all λ cells, challenging the notion that all exchange sites are ionized above a fixed λ threshold.^{68,70–78,80–88} Exchange sites are hydrated non-uniformly, as demonstrated by λ -dependent distributions of Λ values, representing the number of waters within sulfur-

centered ‘inner-spheres’ with 6 Å radii.^{8,40,64,79,89–91} For lower λ cells, the majority of waters reside either within a single exchange site inner-sphere (‘non-bridge’ waters) or multiple inner-spheres (‘bridge’ waters). As λ increases, the proportion of bridge waters decreases, due to the separation of sulfonate groups,^{69,71,76–78,83,84,87} concurrent with the evolution of a water-rich phase composed of outer-sphere waters.

Distributions of α - and β -site Λ values (Λ_α and Λ_β distributions, respectively) with significant overlap were observed. Examination of inner-spheres revealed the complex interplay of three stereoelectronic variables affecting protonation state: (i) the local number density of polar and non-polar fragments, (ii) the spatial arrangement of water, polymer fragments, and ions, and (iii) long-range electrostatic effects. We assert that each exchange site environment yields a characteristic IR spectrum (**premise II**),⁸ and that our Λ_α and Λ_β distributions form the basis for any observed IR spectrum (**premises I, III**).

Taken together, our study focuses on the intrinsic behavior of proton-exchanged Nafion, particularly the interactions between –SO₃H groups and water molecules. By modeling simplified H⁺/H₃O⁺ environments, we systematically explored hydration-dependent proton transfer dynamics, hydrogen-bond structures, and the microstructural organization driven by the acidic side chains. We acknowledge that in real PEM environments, contaminants such as alkali metal cations can influence the acid–base equilibrium and local hydration structure through ionic screening or competitive interactions. While this study aims to isolate and understand the fundamental interactions between –SO₃H groups and water molecules, we reported on the hydration-dependent spectroscopy of metal ion-exchanged Nafion (K⁺, Na⁺, Li⁺, Ca²⁺, Cd²⁺, Ni²⁺, and Al³⁺).^{29,33} Future work can correlate our spectroscopic findings with ReaxFF simulations of metal ion-exchanged Nafion to better understand the impact of external cations.



Intense efforts are focused on the development of environmentally friendly polymer electrolytes that offer high proton conductivity, low gas permeability, and improved water-retention at elevated temperatures. Our methodology is not only applicable to any ionomer, but also to any molecular system undergoing solvation-dependent chemical changes.

Data availability

A collection of files supporting this article are available as part of the ESI† vibrational normal mode animations of model compounds, PDB files of RMD cells, RMD data analysis files, and supporting figures with corresponding discussion. The ESI document provides a description of each supporting file (ESI†).

Author contributions

D. J. D.: conceptualization, data curation, formal analysis, methodology, visualization, and writing – original draft and editing. M. Y. Y.: data curation, formal analysis, investigation, software, writing – MD methods, and writing – review & editing. N. D.: data curation, investigation, supervision, writing – DFT methods, and writing – review & editing. S. S. J.: supervision and writing – review & editing. W. A. G.: conceptualization, project administration, resources, supervision, and writing – review & editing. E. S. S.: conceptualization, funding acquisition, project administration, resources, supervision, writing – review & editing.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

The authors acknowledge financial support from A3 Global (grant code G0007820) and Northeastern University. The ReaxFF work used Expanse at San Diego Supercomputer Center through allocation DMR160114 from the Advanced Cyberinfrastructure Coordination Ecosystem: Services & Support (ACCESS) program, which is supported by National Science Foundation grants #2137603, #2138259, #2138286, #2138296, and #2138307.

References

- 1 United States Department of Energy, *US National Clean Hydrogen Strategy and Roadmap*, 2023.
- 2 United States Department of State; United States Executive Office of the President, *The Long-Term Strategy of the United States: Pathways to Net-Zero Greenhouse Gas Emissions by 2050*, 2021.
- 3 M. El-Shafie, Hydrogen Production by Water Electrolysis Technologies: A Review, *Results Eng.*, 2023, **20**(September), 101426, DOI: [10.1016/j.rineng.2023.101426](https://doi.org/10.1016/j.rineng.2023.101426).
- 4 J. Miyake and K. Miyatake, Fluorine-Free Sulfonated Aromatic Polymers as Proton Exchange Membranes, *Polym. J.*, 2017, **49**(6), 487–495, DOI: [10.1038/pj.2017.11](https://doi.org/10.1038/pj.2017.11).
- 5 M. Adamski, N. Peressin and S. Holdcroft, On the Evolution of Sulfonated Polyphenylenes as Proton Exchange Membranes for Fuel Cells, *Mater. Adv.*, 2021, **2**(15), 4966–5005, DOI: [10.1039/D1MA00511A](https://doi.org/10.1039/D1MA00511A).
- 6 M. Carmo, D. L. Fritz, J. Mergel and D. Stolten, A Comprehensive Review on PEM Water Electrolysis, *Int. J. Hydrogen Energy*, 2013, **38**(12), 4901–4934, DOI: [10.1016/j.ijhydene.2013.01.151](https://doi.org/10.1016/j.ijhydene.2013.01.151).
- 7 United States Department of Energy, *Comparison of Fuel Cell Technologies*, <https://www.energy.gov/eere/fuelcells/comparison-fuel-cell-technologies>, accessed Mar 7, 2024.
- 8 N. Loupe, K. Abu-Hakme, S. Gao, L. Gonzalez, M. Ingargiola, K. Mathiowetz, R. Cruse, J. Doan, A. Schide, I. Salas, N. Dimakis, S. S. Jang, W. A. Goddard and E. S. Smotkin, Group Vibrational Mode Assignments as a Broadly Applicable Tool for Characterizing Ionomer Membrane Structure as a Function of Degree of Hydration, *Chem. Mater.*, 2020, **32**(5), 1828–1843, DOI: [10.1021/acs.chemmater.9b04037](https://doi.org/10.1021/acs.chemmater.9b04037).
- 9 K. A. Mauritz and R. B. Moore, State of Understanding of Nafion, *Chem. Rev.*, 2004, **104**(10), 4535–4586, DOI: [10.1021/cr0207123](https://doi.org/10.1021/cr0207123).
- 10 A. Kusoglu and A. Z. Weber, New Insights into Perfluorinated Sulfonic-Acid Ionomers, *Chem. Rev.*, 2017, **117**(3), 987–1104, DOI: [10.1021/acs.chemrev.6b00159](https://doi.org/10.1021/acs.chemrev.6b00159).
- 11 K.-D. Kreuer, Ion Conducting Membranes for Fuel Cells and Other Electrochemical Devices, *Chem. Mater.*, 2014, **26**(1), 361–380, DOI: [10.1021/cm402742u](https://doi.org/10.1021/cm402742u).
- 12 A. R. Crothers, A. Kusoglu, C. J. Radke and A. Z. Weber, Influence of Mesoscale Interactions on Proton, Water, and Electrokinetic Transport in Solvent-Filled Membranes: Theory and Simulation, *Langmuir*, 2022, **38**(34), 10351–10728, DOI: [10.1021/acs.langmuir.2c00706](https://doi.org/10.1021/acs.langmuir.2c00706).
- 13 M. Soniat and F. A. Houle, How the Hydrophobic Interface between a Perfluorosulfonic Acid Polymer and Water Vapor Controls Membrane Hydration, *ACS Appl. Polym. Mater.*, 2022, **4**(5), 2960–4046, DOI: [10.1021/acsapm.1c01805](https://doi.org/10.1021/acsapm.1c01805).
- 14 M. Fumagalli, S. Lyonnard, G. Prajapati, Q. Berrod, L. Porcar, A. Guillermo and G. Gebel, Fast Water Diffusion and Long-Term Polymer Reorganization during Nafion Membrane Hydration Evidenced by Time-Resolved Small-Angle Neutron Scattering, *J. Phys. Chem. B*, 2015, **119**(23), 7068–7076, DOI: [10.1021/acs.jpcc.5b01220](https://doi.org/10.1021/acs.jpcc.5b01220).
- 15 G. S. Hwang, D. Y. Parkinson, A. Kusoglu, A. A. MacDowell and A. Z. Weber, Understanding Water Uptake and Transport in Nafion Using X-Ray Microtomography, *ACS Macro Lett.*, 2013, **2**(4), 288–291, DOI: [10.1021/mz300651a](https://doi.org/10.1021/mz300651a).
- 16 D. R. Morris and X. Sun, Water-Sorption and Transport Properties of Nafion 117 H, *J. Appl. Polym.*, 1993, **50**(8), 1445–1452.
- 17 J. A. Elliott and S. J. Paddison, Modelling of Morphology and Proton Transport in PFSA Membranes, *Phys. Chem. Chem. Phys.*, 2007, **9**(21), 2602–2618, DOI: [10.1039/b701234a](https://doi.org/10.1039/b701234a).
- 18 H. Rivera, J. S. Lawton, D. E. Budil and E. S. Smotkin, Effect of Sorbed Methanol, Current, and Temperature on Multicomponent Transport in Nafion-Based Direct



- Methanol Fuel Cells, *J. Phys. Chem. B*, 2008, **112**(29), 8542–8548, DOI: [10.1021/jp803158h](#).
- 19 K. Sagarik, M. Phonyiem, C. Lao-ngam and S. Chaiwongwattana, Mechanisms of Proton Transfer in Nafion®: Elementary Reactions at the Sulfonic Acid Groups, *Phys. Chem. Chem. Phys.*, 2008, **10**(15), 2098, DOI: [10.1039/b718480h](#).
 - 20 S. Feng and G. A. Voth, Proton Solvation and Transport in Hydrated Nafion, *J. Phys. Chem. B*, 2011, **115**(19), 5903–5912, DOI: [10.1021/jp2002194](#).
 - 21 R. Devanathan, N. Idupulapati, M. D. Baer, C. J. Mundy and M. Dupuis, Ab Initio Molecular Dynamics Simulation of Proton Hopping in a Model Polymer Membrane, *J. Phys. Chem. B*, 2013, **117**(51), 16522–16529, DOI: [10.1021/jp410229u](#).
 - 22 A. Kusoglu, S. Savagatrup, K. T. Clark and A. Z. Weber, Role of Mechanical Factors in Controlling the Structure–Function Relationship of PFSA Ionomers, *Macromolecules*, 2012, **45**(18), 7467–7476, DOI: [10.1021/ma301419s](#).
 - 23 P. V. Komarov, I. N. Veselov, P. P. Chu and P. G. Khalatur, Mesoscale Simulation of Polymer Electrolyte Membranes Based on Sulfonated Poly(Ether Ether Ketone) and Nafion, *Soft Matter*, 2010, **6**(16), 3939–3956, DOI: [10.1039/b921369d](#).
 - 24 M. Laporta, M. Pegoraro and L. Zanderighi, Perfluorosulfonated Membrane (Nafion): FT-IR Study of the State of Water with Increasing Humidity, *Phys. Chem. Chem. Phys.*, 1999, **1**(19), 4619–4628, DOI: [10.1039/a904460d](#).
 - 25 H. L. Yeager and A. Steck, Cation and Water Diffusion in Nafion Ion Exchange Membranes: Influence of Polymer Structure, *J. Electrochem. Soc.*, 1981, **128**(9), 1880–1884, DOI: [10.1149/1.2127757](#).
 - 26 F. I. Allen, L. R. Comolli, A. Kusoglu, M. A. Modestino, A. M. Minor and A. Z. Weber, Morphology of Hydrated As-Cast Nafion Revealed through Cryo Electron Tomography, *ACS Macro Lett.*, 2015, **4**(1), 1–5, DOI: [10.1021/mz500606h](#).
 - 27 S. Rongpipi, J. M. Chan, A. Bird, G. Freychet, A. Kusoglu and G. M. Su, Revealing Mesoscale Ionomer Membrane Structure by Tender Resonant X-Ray Scattering, *ACS Appl. Polym. Mater.*, 2024, **6**(23), 14115–14123, DOI: [10.1021/acsapm.4c00286](#).
 - 28 J. Doan, E. Kingston, I. Kendrick, K. Anderson, N. Dimakis, P. Knauth, M. L. Di Vona and E. S. Smotkin, Theoretical and Experimental Infrared Spectra of Hydrated and Dehydrated Sulfonated Poly(Ether Ether Ketone), *Polymer*, 2014, **55**(18), 4671–4676, DOI: [10.1016/j.polymer.2014.07.011](#).
 - 29 J. Doan, N. E. Navarro, D. Kumari, K. Anderson, E. Kingston, C. Johnson, A. Vong, N. Dimakis and E. S. Smotkin, Symmetry-Based IR Group Modes as Dynamic Probes of Nafion Ion Exchange Site Structure, *Polymer*, 2015, **73**, 34–41, DOI: [10.1016/j.polymer.2015.07.017](#).
 - 30 M. Webber, N. Dimakis, D. Kumari, M. Fuccillo and E. S. Smotkin, Mechanically Coupled Internal Coordinates of Ionomer Vibrational Modes, *Macromolecules*, 2010, **43**(13), 5500–5502, DOI: [10.1021/ma100915u](#).
 - 31 I. Kendrick, J. Fore, J. Doan, N. Loupe, A. Vong, N. Dimakis, M. Diem and E. S. Smotkin, Operando Raman Micro-Spectroscopy of Polymer Electrolyte Fuel Cells, *J. Electrochem. Soc.*, 2016, **163**(4), H3152–H3159, DOI: [10.1149/2.0211604jes](#).
 - 32 K. Anderson, E. Kingston, J. Romeo, J. Doan, N. Loupe, N. Dimakis and E. S. Smotkin, Infrared Spectroscopy of Ion-Induced Cross-Linked Sulfonated Poly(Ether Ether Ketone), *Polymer*, 2016, **93**, 65–71, DOI: [10.1016/j.polymer.2016.04.015](#).
 - 33 N. Loupe, N. Nasirova, J. Doan, D. Valdez, M. Furlani, N. Dimakis and E. S. Smotkin, DFT - Experimental IR Spectroscopy of Lithiated Single Ion Conducting Perfluorinated Sulfonated Ionomers: Ion Induced Polarization Band Broadening, *J. Electroanal. Chem.*, 2017, **800**, 176–183, DOI: [10.1016/j.jelechem.2017.02.041](#).
 - 34 I. Kendrick, D. Kumari, A. Yakaboski, N. Dimakis and E. S. Smotkin, Elucidating the Ionomer-Electrified Metal Interface, *J. Am. Chem. Soc.*, 2010, **132**(49), 17611–17616, DOI: [10.1021/ja1081487](#).
 - 35 I. Kendrick, A. Yakaboski, E. Kingston, J. Doan, N. Dimakis and E. S. Smotkin, Theoretical and Experimental Infrared Spectra of Hydrated and Dehydrated Nafion, *J. Polym. Sci., Part B: Polym. Phys.*, 2013, **51**(18), 1329–1334, DOI: [10.1002/polb.23348](#).
 - 36 E. Kraka, M. Quintano, H. W. La Force, J. J. Antonio and M. Freindorf, The Local Vibrational Mode Theory and Its Place in the Vibrational Spectroscopy Arena, *J. Phys. Chem. A*, 2022, **126**(47), 8781–8798, DOI: [10.1021/acs.jpca.2c05962](#).
 - 37 M. Yamaguchi and A. Ohira, Vibrational Analysis of Side Chain Model Compounds of Perfluorinated Alkyl Sulfonic Acid Ionomers, *J. Phys. Chem. A*, 2012, **116**(44), 10850–10863, DOI: [10.1021/jp308663c](#).
 - 38 A. Moran and S. Mukamel, The Origin of Vibrational Mode Couplings in Various Secondary Structural Motifs of Polypeptides, *Proc. Natl. Acad. Sci. U. S. A.*, 2004, **101**(2), 506–510, DOI: [10.1073/pnas.2533089100](#).
 - 39 E. Kraka, W. Zou and Y. Tao, Decoding Chemical Information from Vibrational Spectroscopy Data: Local Vibrational Mode Theory, *Wiley Interdiscip. Rev.: Comput. Mol. Sci.*, 2020, **10**(5), 1–34, DOI: [10.1002/wcms.1480](#).
 - 40 N. J. Silva, D. Tunega, C. Korzeniewski, H. Lischka and A. J. A. Aquino, Microhydration of Polymer Electrolyte Membranes: A Comparison of Hydrogen-Bonding Networks and Spectral Properties of Nafion and Bis [(Perfluoroalkyl)Sulfonyl] Imide, *J. Phys. Chem. B*, 2019, **123**(46), 9899–9911, DOI: [10.1021/acs.jpcc.9b07815](#).
 - 41 Y. Liang, J. P. Kitt, S. D. Minter, J. M. Harris and C. Korzeniewski, Vibrational Spectroscopic Monitoring of the Gelation Transition in Nafion Ionomer Dispersions, *Appl. Spectrosc.*, 2021, **75**(4), 376–384, DOI: [10.1177/0003702820949129](#).
 - 42 M. A. Barique, E. Tsuchida, A. Ohira and K. Tashiro, Effect of Elevated Temperatures on the States of Water and Their



- Correlation with the Proton Conductivity of Nafion, *ACS Omega*, 2018, 3(1), 349–360, DOI: [10.1021/acsomega.7b01765](#).
- 43 R. K. Singh, K. Kunimatsu, K. Miyatake and T. Tsuneda, Experimental and Theoretical Infrared Spectroscopic Study on Hydrated Nafion Membrane, *Macromolecules*, 2016, 49(17), 6621–6629, DOI: [10.1021/acs.macromol.6b00999](#).
 - 44 C. K. Byun, T. Parker, C. Liang, I. Kendrick, N. Dimakis, E. S. Smotkin, L. M. Jin, D. Zhuang, D. D. Desmarteau, S. E. Creager and C. Korzeniewski, Thermal Processing as a Means to Prepare Durable, Submicron Thickness Ionomer Films for Study by Transmission Infrared Spectroscopy, *Anal. Chem.*, 2012, 84(19), 8127–8132, DOI: [10.1021/ac301662f](#).
 - 45 M. Danilczuk, L. Lin, S. Schlick, S. J. Hamrock and M. S. Schaberg, Understanding the Fingerprint Region in the Infra-Red Spectra of Perfluorinated Ionomer Membranes and Corresponding Model Compounds: Experiments and Theoretical Calculations, *J. Power Sources*, 2011, 196(20), 8216–8224, DOI: [10.1016/j.jpowsour.2011.05.067](#).
 - 46 D. S. Warren and A. J. McQuillan, Infrared Spectroscopic and DFT Vibrational Mode Study of Perfluoro(2-Ethoxyethane) Sulfonic Acid (PES), a Model Nafion Side-Chain Molecule, *J. Phys. Chem. B*, 2008, 112(34), 10535–10543, DOI: [10.1021/jp801838n](#).
 - 47 T. Shimoaka, C. Wakai, T. Sakabe, S. Yamazaki and T. Hasegawa, Hydration Structure of Strongly Bound Water on the Sulfonic Acid Group in a Nafion Membrane Studied by Infrared Spectroscopy and Quantum Chemical Calculation, *Phys. Chem. Chem. Phys.*, 2015, 17(14), 8843–8849, DOI: [10.1039/c5cp00567a](#).
 - 48 M. Ludvigsson, J. Lindgren and J. Tegenfeldt, FTIR Study of Water in Cast Nafion Films, *Electrochim. Acta*, 2000, 45(14), 2267–2271, DOI: [10.1016/S0013-4686\(99\)00438-7](#).
 - 49 S. D. Bernardina, J.-B. Brubach, Q. Berrod, A. Guillermo, P. Judeinstein, P. Roy and S. Lyonnard, Mechanism of Ionization, Hydration, and Intermolecular H-Bonding in Proton Conducting Nanostructured Ionomers, *J. Phys. Chem. C*, 2014, 118(44), 25468–25479, DOI: [10.1021/jp5074818](#).
 - 50 K. Feng, L. Hou, B. Tang and P. Wu, Does Thermal Treatment Merely Make a H₂O-Saturated Nafion Membrane Lose Its Absorbed Water at High Temperature?, *Phys. Chem. Chem. Phys.*, 2015, 17(14), 9106–9115, DOI: [10.1039/c5cp00203f](#).
 - 51 R. Basnayake, G. R. Peterson, D. J. Casadonte and C. Korzeniewski, Hydration and Interfacial Water in Nafion Membrane Probed by Transmission Infrared Spectroscopy, *J. Phys. Chem. B*, 2006, 110(47), 23938–23943, DOI: [10.1021/jp064121i](#).
 - 52 J. Ostrowska and A. Narebska, Infrared Study of Hydration and Association of Functional Groups in a Perfluorinated Nafion Membrane, Part 1, *Colloid Polym. Sci.*, 1983, 261(2), 93–98, DOI: [10.1007/BF01410686](#).
 - 53 V.-A. Glezakou, M. Dupuis and C. J. Mundy, Acid/Base Equilibria in Clusters and Their Role in Proton Exchange Membranes: Computational Insight, *Phys. Chem. Chem. Phys.*, 2007, 9(43), 5752, DOI: [10.1039/b709752b](#).
 - 54 M. Phonyiem, S. Chaiwongwattana, C. Lao-ngam and K. Sagarik, Proton Transfer Reactions and Dynamics of Sulfonic Acid Group in Nafion®, *Phys. Chem. Chem. Phys.*, 2011, 13(23), 10923–10939, DOI: [10.1039/c1cp20469f](#).
 - 55 S. J. Paddison, K. D. Kreuer and J. Maier, About the Choice of the Protogenic Group in Polymer Electrolyte Membranes: Ab Initio Modelling of Sulfonic Acid, Phosphonic Acid, and Imidazole Functionalized Alkanes, *Phys. Chem. Chem. Phys.*, 2006, 8(39), 4530–4542, DOI: [10.1039/b611221h](#).
 - 56 T. E. Springer, T. A. Zawodzinski and S. Gottesfeld, Polymer Electrolyte Fuel Cell Model, *J. Electrochem. Soc.*, 1991, 138(8), 2334–2342, DOI: [10.1149/1.2085971](#).
 - 57 S. J. Paddison and J. A. Elliott, On the Consequences of Side Chain Flexibility and Backbone Conformation on Hydration and Proton Dissociation in Perfluorosulfonic Acid Membranes, *Phys. Chem. Chem. Phys.*, 2006, 8(18), 2193–2203, DOI: [10.1039/b602188c](#).
 - 58 S. J. Paddison and J. A. Elliott, Molecular Modeling of the Short-Side-Chain Perfluorosulfonic Acid Membrane, *J. Phys. Chem. A*, 2005, 109(33), 7583–7593, DOI: [10.1021/jp0524734](#).
 - 59 M. A. Ilhan and E. Spohr, Ab Initio Molecular Dynamics of Proton Networks in Narrow Polymer Electrolyte Pores, *J. Phys. Condens. Matter*, 2011, 23(23), 234104, DOI: [10.1088/0953-8984/23/23/234104](#).
 - 60 H. Sakai and T. Tokumasu, Reaction Analysis for Deprotonation of the Sulfonic Group of Perfluorosulfonic Acid Molecules at Low Hydration Levels, *J. Phys. Chem. A*, 2014, 118(1), 275–282, DOI: [10.1021/jp409781s](#).
 - 61 F. Sepehr and S. J. Paddison, Primary Hydration and Proton Transfer of Electrolyte Acids: An Ab Initio Study, *Solid State Ionics*, 2017, 306, 2–12, DOI: [10.1016/j.ssi.2017.03.013](#).
 - 62 S. Singh, T. Taketsugu and R. K. Singh, Hydration, Prediction of the PKa, and Infrared Spectroscopic Study of Sulfonated Polybenzophenone (SPK) Block-Copolymer Hydrocarbon Membranes and Comparisons with Nafion, *ACS Omega*, 2021, 6(48), 32739–32748, DOI: [10.1021/acsomega.1c04484](#).
 - 63 R. K. Singh, T. Tsuneda, K. Miyatake and M. Watanabe, Theoretical Investigation of Local Proton Conductance in the Proton Exchange Membranes, *Chem. Phys. Lett.*, 2014, 608, 11–16, DOI: [10.1016/j.cplett.2014.05.076](#).
 - 64 T. Zelovich, K. I. Winey and M. E. Tuckerman, Hydronium Ion Diffusion in Model Proton Exchange Membranes at Low Hydration: Insights from Ab Initio Molecular Dynamics, *J. Mater. Chem. A*, 2021, 9(4), 2448–2458, DOI: [10.1039/d0ta10565a](#).
 - 65 B. F. Habenicht, S. J. Paddison and M. E. Tuckerman, The Effects of the Hydrophobic Environment on Proton Mobility in Perfluorosulfonic Acid Systems: An Ab Initio Molecular Dynamics Study, *J. Mater. Chem.*, 2010, 20(30), 6342–6351, DOI: [10.1039/c0jm00253d](#).
 - 66 C. Wang, J. K. Clark, M. Kumar and S. J. Paddison, An Ab Initio Study of the Primary Hydration and Proton Transfer



- of CF₃SO₃H and CF₃O(CF₂)₂SO₃H: Effects of the Hybrid Functional and Inclusion of Diffuse Functions, *Solid State Ionics*, 2011, **199–200**(1), 6–13, DOI: [10.1016/j.ssi.2011.07.002](#).
- 67 S. J. Paddison, The Modeling of Molecular Structure and Ion Transport in Sulfonic Acid Based Ionomer Membranes, *J. New Mater. Electrochem. Syst.*, 2001, **4**, 197–207.
- 68 T. Flottat, B. Latour, F. Goujon, P. Hauret and P. Malfreyt, Investigating Percolation and Clustering Effects on Aquivion and Nafion Membranes at the Molecular Scale, *Int. J. Hydrogen Energy*, 2023, **48**(85), 33283–33296, DOI: [10.1016/j.ijhydene.2023.05.086](#).
- 69 R. Lawler, C. Caliendo, H. Ju, J. Y. Kim, S. W. Lee and S. S. Jang, Effect of the Side-Chain Length in Perfluorinated Sulfonic and Phosphoric Acid-Based Membranes on Nanophase Segregation and Transport: A Molecular Dynamics Simulation Approach, *J. Phys. Chem. B*, 2020, **124**(8), 1571–1580, DOI: [10.1021/acs.jpcc.9b10408](#).
- 70 S. S. Jang, V. Molinero, T. Çağın and W. A. Goddard, Nanophase-Segregation and Transport in Nafion 117 from Molecular Dynamics Simulations: Effect of Monomeric Sequence, *J. Phys. Chem. B*, 2004, **108**(10), 3149–3157, DOI: [10.1021/jp036842c](#).
- 71 A. T. Kuo, W. Shinoda and S. Okazaki, Molecular Dynamics Study of the Morphology of Hydrated Perfluorosulfonic Acid Polymer Membranes, *J. Phys. Chem. C*, 2016, **120**(45), 25832–25842, DOI: [10.1021/acs.jpcc.6b08015](#).
- 72 D. W. M. Hofmann, L. Kuleshova, B. D'Aguzzo, V. D. Noto, E. Negro, F. Conti and M. Vittadello, Investigation of Water Structure in Nafion Membranes by Infrared Spectroscopy and Molecular Dynamics Simulation, *J. Phys. Chem. B*, 2009, **113**(3), 632–639, DOI: [10.1021/jp0806761h](#).
- 73 R. Wang, S. Liu, L. Wang, M. Li and C. Gao, Understanding of Nanophase Separation and Hydrophilic Morphology in Nafion and SPEEK Membranes: A Combined Experimental and Theoretical Studies, *Nanomaterials*, 2019, **9**(6), 869, DOI: [10.3390/nano9060869](#).
- 74 Y. Zhao, Q. Gao, X. Xu, C. Ma, Q. He, Y. Min and S. Zhao, *Compromise Mechanism of Proton Transfer in Crown Ether-Based Biomimetic Proton Exchange Membranes: Insights from Molecular Dynamics Simulations*, Elsevier B.V., 2025, vol. 715, DOI: [10.1016/j.memsci.2024.123456](#).
- 75 S. Akbari, M. T. Hamed Mosavian, F. Moosavi and A. Ahmadvour, Elucidating the Morphological Aspects and Proton Dynamics in a Hybrid Perfluorosulfonic Acid Membrane for Medium-Temperature Fuel Cell Applications, *Phys. Chem. Chem. Phys.*, 2018, **20**(47), 29778–29789, DOI: [10.1039/C8CP05377D](#).
- 76 S. Cui, J. Liu, M. E. Selvan, S. J. Paddison, D. J. Keffer and B. J. Edwards, Comparison of the Hydration and Diffusion of Protons in Perfluorosulfonic Acid Membranes with Molecular Dynamics Simulations, *J. Phys. Chem. B*, 2008, **112**(42), 13273–13284, DOI: [10.1021/jp8039803](#).
- 77 S. Cui, J. Liu, M. E. Selvan, D. J. Keffer, B. J. Edwards and W. V. Steele, A Molecular Dynamics Study of a Nafion Polyelectrolyte Membrane and the Aqueous Phase Structure for Proton Transport, *J. Phys. Chem. B*, 2007, **111**(9), 2208–2218, DOI: [10.1021/jp066388n](#).
- 78 A. Venkatnathan, R. Devanathan and M. Dupuis, Atomistic Simulations of Hydrated Nafion and Temperature Effects on Hydronium Ion Mobility, *J. Phys. Chem. B*, 2007, **111**(25), 7234–7244, DOI: [10.1021/jp0700276](#).
- 79 S. Sengupta and A. V. Lyulin, Molecular Modeling of Structure and Dynamics of Nafion Protonation States, *J. Phys. Chem. B*, 2019, **123**(31), 6882–6891, DOI: [10.1021/acs.jpcc.9b04534](#).
- 80 D. W. M. Hofmann, L. Kuleshova and B. D'Aguzzo, Molecular Dynamics Simulation of Hydrated Nafion with a Reactive Force Field for Water, *J. Mol. Model.*, 2008, **14**(3), 225–235, DOI: [10.1007/s00894-007-0265-9](#).
- 81 R. Cui, S. Li, C. Yu and Y. Zhou, The Evolution of Hydrogen Bond Network in Nafion via Molecular Dynamics Simulation, *Macromolecules*, 2023, **56**(4), 1688–1703, DOI: [10.1021/acs.macromol.2c02106](#).
- 82 A. Rahbari, R. Hartkamp, O. A. Moulto, A. Bos, L. J. P. van den Broeke, M. Ramdin, D. Dubbeldam, A. V. Lyulin and T. J. H. Vlugt, Electro-Osmotic Drag and Thermodynamic Properties of Water in Hydrated Nafion Membranes from Molecular Dynamics, *J. Phys. Chem. C*, 2022, **126**(18), 8121–8133, DOI: [10.1021/acs.jpcc.2c01226](#).
- 83 M. Bazaid, Y. Huang, W. A. Goddard and S. S. Jang, Proton Transport through Interfaces in Nanophase-Separation of Hydrated Aquivion Membrane: Molecular Dynamics Simulation Approach, *Colloids Surf. A Physicochem. Eng. Asp.*, 2023, **676**, 1–11, DOI: [10.1016/j.colsurfa.2023.132187](#).
- 84 S. Urata, J. Irisawa, A. Takada, W. Shinoda, S. Tsuzuki and M. Mikami, Molecular Dynamics Simulation of Swollen Membrane of Perfluorinated Ionomer, *J. Phys. Chem. B*, 2005, **109**(9), 4269–4278, DOI: [10.1021/jp046434o](#).
- 85 R. Devanathan, A. Venkatnathan and M. Dupuis, Atomistic Simulation of Nafion Membrane. 2. Dynamics of Water Molecules and Hydronium Ions, *J. Phys. Chem. B*, 2007, **111**(45), 13006–13013, DOI: [10.1021/jp0761057](#).
- 86 R. Devanathan, A. Venkatnathan, R. Rousseau, M. Dupuis, T. Frigato, W. Gu and V. Helms, Atomistic Simulation of Water Percolation and Proton Hopping in Nafion Fuel Cell Membrane, *J. Phys. Chem. B*, 2010, **114**(43), 13681–13690, DOI: [10.1021/jp103398b](#).
- 87 R. Devanathan, A. Venkatnathan and M. Dupuis, Atomistic Simulation of Nafion Membrane: I. Effect of Hydration on Membrane Nanostructure, *J. Phys. Chem. B*, 2007, **111**(28), 8069–8079, DOI: [10.1021/jp0726992](#).
- 88 A. V. Lyulin, S. Sengupta, A. Varughese, P. Komarov and A. Venkatnathan, Effect of Annealing on Structure and Diffusion in Hydrated Nafion Membranes, *ACS Appl. Polym. Mater.*, 2020, **2**(11), 5058–5066, DOI: [10.1021/acsapm.0c00875](#).
- 89 S. J. Paddison and J. A. Elliott, Selective Hydration of the 'Short-Side-Chain' Perfluorosulfonic Acid Membrane. An



- ONIOM Study, *Solid State Ionics*, 2007, **178**(7–10), 561–567, DOI: [10.1016/j.ssi.2007.01.007](#).
- 90 M. Wang, K. W. Feindel, S. H. Bergens and R. E. Wasylshen, In Situ Quantification of the In-Plane Water Content in the Nafion® Membrane of an Operating Polymer-Electrolyte Membrane Fuel Cell Using ¹H Micro-Magnetic Resonance Imaging Experiments, *J. Power Sources*, 2010, **195**(21), 7316–7322, DOI: [10.1016/j.jpowsour.2010.05.029](#).
- 91 Z. Peng, V. Badets, P. Huguet, A. Morin, P. Schott, T. B. H. Tran, M. Porozhnyy, V. Nikonenko and S. Deabate, Operando μ -Raman Study of the Actual Water Content of Perfluorosulfonic Acid Membranes in the Fuel Cell, *J. Power Sources*, 2017, **356**, 200–211, DOI: [10.1016/j.jpowsour.2017.04.095](#).
- 92 A. C. T. van Duin, S. Dasgupta, F. Lorant and W. A. Goddard, ReaxFF: A Reactive Force Field for Hydrocarbons, *J. Phys. Chem. A*, 2001, **105**(41), 9396–9409, DOI: [10.1021/jp004368u](#).
- 93 T. P. Senftle, S. Hong, M. M. Islam, S. B. Kylasa, Y. Zheng, Y. K. Shin, C. Junkermeier, R. Engel-Herbert, M. J. Janik, H. M. Aktulga, T. Verstraelen, A. Grama and A. C. T. Van Duin, The ReaxFF Reactive Force-Field: Development, Applications and Future Directions, *npj Comput. Mater.*, 2016, **2**, 15011, DOI: [10.1038/npjcompumats.2015.11](#).
- 94 T. Mabuchi and T. Tokumasu, Molecular Dynamics Simulation of Proton Transport in Polymer Electrolyte Membrane, *J. Nanosci. Nanotechnol.*, 2015, **15**(4), 2958–2963, DOI: [10.1166/jnn.2015.9647](#).
- 95 T. Mabuchi and T. Tokumasu, Relationship between Proton Transport and Morphology of Perfluorosulfonic Acid Membranes: A Reactive Molecular Dynamics Approach, *J. Phys. Chem. B*, 2018, **122**(22), 5922–5932, DOI: [10.1021/acs.jpcc.8b02318](#).
- 96 M. K. Petersen and G. A. Voth, Characterization of the Solvation and Transport of the Hydrated Proton in the Perfluorosulfonic Acid Membrane Nafion, *J. Phys. Chem. B*, 2006, **110**(37), 18594–18600, DOI: [10.1021/jp062719k](#).
- 97 D. Seeliger, C. Hartnig and E. Spohr, Aqueous Pore Structure and Proton Dynamics in Solvated Nafion Membranes, *Electrochim. Acta*, 2005, **50**(21), 4234–4240, DOI: [10.1016/j.electacta.2005.03.071](#).
- 98 Y.-L. S. Tse, A. M. Herring, K. Kim and G. A. Voth, Molecular Dynamics Simulations of Proton Transport in 3M and Nafion Perfluorosulfonic Acid Membranes, *J. Phys. Chem. C*, 2013, **117**(16), 8079–8091, DOI: [10.1021/jp400693g](#).
- 99 Y. K. Choe, E. Tsuchida, T. Ikeshoji, S. Yamakawa and S. A. Hyodo, Nature of Water Transport and Electro-Osmosis in Nafion: Insights from First-Principles Molecular Dynamics Simulations under an Electric Field, *J. Phys. Chem. B*, 2008, **112**(37), 11586–11594, DOI: [10.1021/jp8041878](#).
- 100 Y. K. Choe, E. Tsuchida, T. Ikeshoji, S. Yamakawa and S. A. Hyodo, Nature of Proton Dynamics in a Polymer Electrolyte Membrane, Nafion: A First-Principles Molecular Dynamics Study, *Phys. Chem. Chem. Phys.*, 2009, **11**(20), 3892–3899, DOI: [10.1039/b819535h](#).
- 101 D. I. Bower and W. F. Maddams, *The Vibrational Spectroscopy of Polymers*, Cambridge University Press, Cambridge, 1989.
- 102 S. Grimme, J. Antony, S. Ehrlich and H. Krieg, A Consistent and Accurate Ab Initio Parametrization of Density Functional Dispersion Correction (DFT-D) for the 94 Elements H–Pu, *J. Chem. Phys.*, 2010, **132**(15), 154104, DOI: [10.1063/1.3382344](#).
- 103 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**(18), 3865–3868, DOI: [10.1103/PhysRevLett.77.3865](#).
- 104 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple [Phys. Rev. Lett. 77, 3865 (1996)], *Phys. Rev. Lett.*, 1997, **78**(7), 1396, DOI: [10.1103/PhysRevLett.78.1396](#).
- 105 C. Adamo and V. Barone, Toward Reliable Density Functional Methods without Adjustable Parameters: The PBE0 Model, *J. Chem. Phys.*, 1999, **110**(13), 6158–6170, DOI: [10.1063/1.478522](#).
- 106 M. J. Frisch, J. A. Pople and J. S. Binkley, Self-Consistent Molecular Orbital Methods 25. Supplementary Functions for Gaussian Basis Sets, *J. Chem. Phys.*, 1984, **80**(7), 3265–3269, DOI: [10.1063/1.447079](#).
- 107 T. Clark, J. Chandrasekhar, G. W. Spitznagel and P. V. R. Schleyer, Efficient Diffuse Function-augmented Basis Sets for Anion Calculations. III. The 3-21+G Basis Set for First-row Elements, Li–F, *J. Comput. Chem.*, 1983, **4**(3), 294–301, DOI: [10.1002/jcc.540040303](#).
- 108 L. Martínez, R. Andrade, E. G. Birgin and J. M. Martínez, Packmol: A Package for Building Initial Configurations for Molecular Dynamics Simulations, *J. Comput. Chem.*, 2009, **30**(13), 2157–2164, DOI: [10.1002/jcc.21224](#).
- 109 R. Devanathan, N. Idupulapati, M. D. Baer, C. J. Mundy and M. Dupuis, Ab Initio Molecular Dynamics Simulation of Proton Hopping in a Model Polymer Membrane, *J. Phys. Chem. B*, 2013, **117**(51), 16522–16529, DOI: [10.1021/jp410229u](#).
- 110 A. K. Rappé, C. J. Casewit, K. S. Colwell, I. Goddard and W. M. Skiff, UFF, a Full Periodic Table Force Field for Molecular Mechanics and Molecular Dynamics Simulations, *J. Am. Chem. Soc.*, 1992, **114**(25), 10024–10035, DOI: [10.1021/ja00051a040](#).
- 111 S. Plimpton, Fast Parallel Algorithms for Short-Range Molecular Dynamics, *J. Comput. Phys.*, 1995, **117**(1), 1–19, DOI: [10.1006/jcph.1995.1039](#).
- 112 A. P. Thompson, H. M. Aktulga, R. Berger, D. S. Bolintineanu, W. M. Brown, P. S. Crozier, P. J. in't Veld, A. Kohlmeyer, S. G. Moore, T. D. Nguyen, R. Shan, M. J. Stevens, J. Tranchida, C. Trott and S. J. Plimpton, LAMMPS - a Flexible Simulation Tool for Particle-Based Materials Modeling at the Atomic, Meso, and Continuum Scales, *Comput. Phys. Commun.*, 2022, **271**, 108171, DOI: [10.1016/j.cpc.2021.108171](#).



- 113 L. Verlet, Computer “Experiments” on Classical Fluids. I. Thermodynamical Properties of Lennard-Jones Molecules, *Phys. Rev.*, 1967, **159**(98), 98–103, DOI: [10.1103/PhysRev.159.98](#).
- 114 R. W. Hockney and J. W. Eastwood, *Computer Simulation Using Particles*, McGraw-Hill International Book Company, New York, 1981.
- 115 A. K. Rappe and I. Goddard, Charge Equilibration for Molecular Dynamics Simulations, *J. Phys. Chem.*, 1991, **95**(8), 3358–3363, DOI: [10.1021/j100161a070](#).
- 116 S. J. Paddison, D. W. Reagor and T. A. Zawodzinski Jr, High Frequency Dielectric Studies of Hydrated Nafion, *J. Electroanal. Chem.*, 1998, **459**(1), 91–97, DOI: [10.1016/S0022-0728\(98\)00321-0](#).
- 117 H. A. Laitinen and W. E. Harris, *Chemical Analysis*, ed. R. H. Summersgill and S. L. Langman, McGraw-Hill, Inc., New York, 2nd edn, 1975.
- 118 M. B. Herath, S. E. Creager, A. Kitaygorodskiy and D. D. DesMarteau, Perfluoroalkyl Phosphonic and Phosphinic Acids as Proton Conductors for Anhydrous Proton-Exchange Membranes, *ChemPhysChem*, 2010, **11**(13), 2871–2878, DOI: [10.1002/cphc.201000184](#).
- 119 M. Schuster, T. Rager, A. Noda, K. D. Kreuer and J. Maier, About the Choice of the Protogenic Group in PEM Separator Materials for Intermediate Temperature, Low Humidity Operation: A Critical Comparison of Sulfonic Acid, Phosphonic Acid and Imidazole Functionalized Model Compounds, *Fuel Cells*, 2005, **5**(3), 355–365, DOI: [10.1002/fuce.200400059](#).
- 120 P. A. H. Wyatt, Sulphuric Acid Autoprotolysis and the Theory of Dissociated Solvents, *Trans. Faraday Soc.*, 1969, **65**, 585–590, DOI: [10.1039/tf9696500585](#).
- 121 R. H. Flowers, R. J. Gillespie and E. A. Robinson, The Sulphuric Acid Solvent System: Part I. Acid-base Reactions, *Can. J. Chem.*, 1960, **38**(8), 1363–1370, DOI: [10.1139/v60-190](#).
- 122 R. Buzzoni, S. Bordiga, G. Ricchiardi, G. Spoto and A. Zecchina, Interaction of H₂O, CH₃OH, (CH₃)₂O, CH₃CN, and Pyridine with the Superacid Perfluorosulfonic Membrane Nafion: An IR and Raman Study, *J. Phys. Chem.*, 1995, **99**(31), 11937–11951, DOI: [10.1021/j100031a023](#).
- 123 E. Negro, M. Vittadello, K. Vezzù, S. J. Paddison and V. Di Noto, The Influence of the Cationic Form and Degree of Hydration on the Structure of Nafion™, *Solid State Ionics*, 2013, **252**, 84–92, DOI: [10.1016/j.ssi.2013.09.017](#).
- 124 A. Vishnyakov and A. V. Neimark, Molecular Dynamics Simulation of Nafion Oligomer Solvation in Equimolar Methanol–Water Mixture, *J. Phys. Chem. B*, 2001, **105**(32), 7830–7834, DOI: [10.1021/jp004082p](#).

