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## The remarkable performance of a single iridium atom supported on hematite for methane activation: a density functional theory study<sup>†</sup>

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Methane is the major component of natural gas, and it significantly contributes to global warming. In this study, we investigated methane activation on the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface and M/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surfaces (M = Ag, Ir, Cu, or Co) using the density-functional theory (DFT) + *U* method. Our study shows that the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface is a more effective catalyst for C–H bond activation than other catalyst surfaces. We have applied electron density difference (EDD), density of states (DOS), and Bader charge calculations to confirm the cooperative CH···O and agostic interactions between CH<sub>4</sub> and the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface. To further modify the reactivity of the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface towards methane activation, we conducted a study of the effect of oxygen vacancy (O<sub>V</sub>) on C–H activation and CH<sub>4</sub> dehydrogenation. In the comparison of pristine  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), and Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surfaces, the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is the best in terms of CH<sub>4</sub> adsorption energy and C–H bond elongation, whereas the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface catalyst has the lowest C–H bond activation barrier for the CH<sub>4</sub> molecule. The calculations indicate that the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface could be a candidate catalyst for CH<sub>4</sub> dehydrogenation reactions.

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## 1. Introduction

Methane is the major component of natural gas and a potential clean energy carrier because of its highest H to C ratio of any fossil fuel.<sup>1–4</sup> With the industrial revolution, methane is considered an alternative to nonrenewable petroleum resources and plays a significant role in the energy field.<sup>5–8</sup> Consequently, the combustion of methane leads to 25% of the global CO<sub>2</sub> emission, and methane itself is a greenhouse gas, which has a relative greenhouse impact of 25 times greater than CO<sub>2</sub>. This rise in the greenhouse gas concentration, predominantly methane in the earth's atmosphere, triggered an increase in global air pollution and surface temperature.<sup>9,10</sup> Hence, the methane conversion to valuable commodity chemicals near its source is essential in the industry and addresses environmental concerns.

Over the decades, various strategies, including direct and indirect routes, have been developed to use methane effectively;

however, the C–H bond activation under mild conditions is challenging.<sup>10–13</sup> The chemical products based on direct methane conversion have received significant interest in recent decades from industry and academia due to their more energy-efficient and environmental-friendliness.<sup>14–16</sup> The rutile IrO<sub>2</sub> surface has been the subject of intensive theoretical<sup>17–19</sup> and experimental studies,<sup>20–22</sup> among the different catalysts in methane conversion owing to its remarkable reactivity. In addition, much evidence endorses this superior catalytic reactivity and intrinsic electronic characteristics of the IrO<sub>2</sub> surface in methane activation.<sup>23</sup> Besides its higher reactivity, the high cost and the natural scarcity of iridium significantly limit its further development and application.

The doping of a single metal atom in metal oxide surfaces has attracted considerable interest, as it could afford precious metal thrifing and solve the ultimate goal of designing efficient and low-cost catalysts in heterogeneous catalysis.<sup>24–28</sup> For example, Y. Meng *et al.* demonstrated that the doping of the Pd atom improved the thermodynamic stability and catalytic performance of the Cu(111) surface towards partial oxidation of methane than the single Pd atom adsorbed surface.<sup>28</sup> Similarly, Guo and coworkers<sup>29</sup> have recently shown that Rh doped Ni(111) catalyst exhibits promising performance for coke resistance in the CH<sub>4</sub>/CO<sub>2</sub> reforming reaction. Besides, the density functional theory calculations by Eisenberg and Baer have demonstrated that doping of Li atom on MgO significantly eases the hydrogen abstraction reaction from methane.<sup>27</sup> Furthermore, recent

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studies have described the role of single metal atom dopants in  $\text{Al}_2\text{O}_3$  and zeolites for the activity, selectivity, stability, and regeneration capability of catalysts in propane and light alkane dehydrogenation.<sup>30-32</sup> Previous studies have shown that the metal oxides doped with low-valence metal atoms show better

catalytic activity towards methane activation than the non-doped oxides.<sup>33-35</sup> Compared with other metal oxides, hematite is abundant, inexpensive, and environmentally benign. It has been widely used as a catalyst and supports catalysis because of its stability and attractive physicochemical

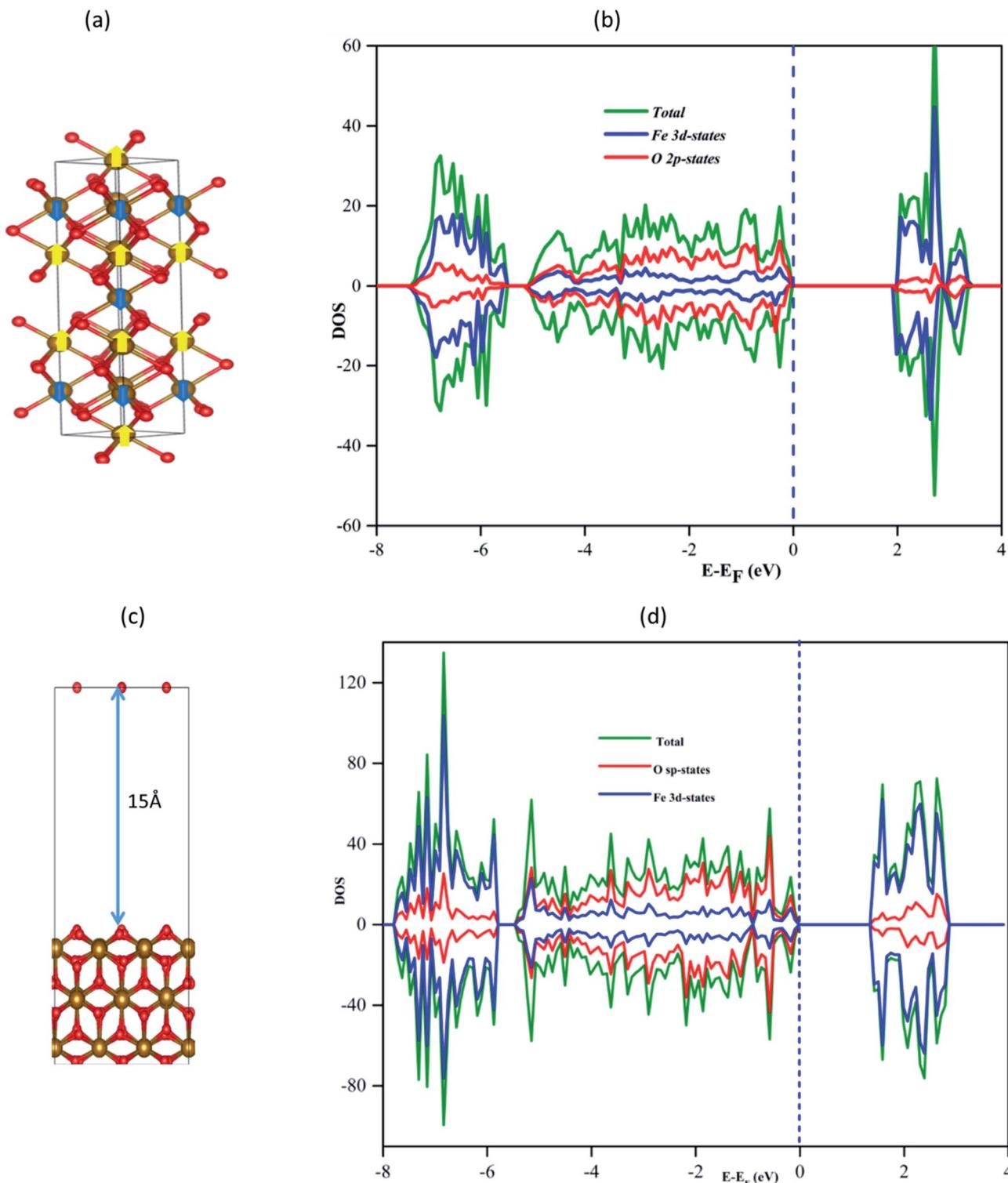


Fig. 1 (a)  $\alpha\text{-Fe}_2\text{O}_3$  lattice with an antiferromagnetic spin arrangement indicated; spin up by yellow and down spin by blue arrows at Fe sites (b) density of states of the bulk hematite (c) calculated optimum hematite (110) surface structure and (d) DOS of the  $\alpha\text{-Fe}_2\text{O}_3$ (110) surface plane.

properties.<sup>36–38</sup> Furthermore currently researchers used iridium oxide as a catalyst for methane activation. But iridium oxide is rare earth metal oxide and it is too expensive.

Motivated by these analyses and to design the low-cost catalyst with the comparable catalytic performance of  $\text{IrO}_2(110)$  surface, in this work, we designed a distinct type of different single-atom doped on hematite ( $\alpha\text{-Fe}_2\text{O}_3$ ) surface. In addition, we explored its catalytic activity towards methane adsorption and activation using density functional theory calculations. Furthermore, previous studies<sup>39</sup> indicate that the presence of oxygen vacancies in the metal oxide supports enhanced the methane activation; hence we also examined the effects of a single oxygen vacancy on the hematite support for C–H activation.

## 2. Computational details

All the spin-polarized DFT calculations were carried out using the widely used plane wave Vienna *ab initio* simulation package (VASP).<sup>40,41</sup> The nonlocal optB88-vdW exchange–correlation functional was used in this study, which explicitly includes the effects of van der Waals forces.<sup>42–44</sup> The core electrons were represented by the projector augmented wave (PAW),<sup>45</sup> and the energy cutoff for all the calculations was set to 500 eV based on our benchmark results. The atomic positions were optimized until the total energy converged to  $10^{-4}$  eV, and the atomic positions were relaxed using the quasi-Newton algorithm until the  $x$ ,  $y$ , and  $z$  components of the unconstrained atomic force were smaller  $2 \times 10^{-2}$  eV  $\text{\AA}^{-1}$ . To treat the on-site Coulomb and exchange interaction of the strongly localized 3d electrons of metal atoms, here we employed the DFT +  $U$  mechanism. The DFT +  $U$  is a widely used method to describe transition metal-related studies and we conduct the benchmark calculation of the  $U$  value for each transition metal. Thus, one of the most common approaches to determine the appropriate values of  $U$  is to compare the calculated band gap for a set of  $U$  values with the experimental band gap.<sup>46,47</sup> As shown in Fig. 1S,† when the  $U$  value increases the band gap of the oxide also increase for all dopant case. The experimental band gaps are 0,<sup>48</sup> 2.18,<sup>49</sup> 2.85,<sup>50</sup> 1.6,<sup>51</sup> and 1.3 eV<sup>52</sup> for  $\text{IrO}_2$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{CuO}$ ,  $\text{CoO}$ , and  $\text{Ag}_2\text{O}$ , respectively. Therefore, we use a combination of calculated  $U$  values derived from the fitting band gaps. For Ag, Cu, Ir, Co, and Fe, we use 5.0 eV, 4.0 eV, 0.0 eV, 3.0 eV, and 4.0 eV, respectively. Similar studies also use combination  $U$  values for heteroatom systems.<sup>53–57</sup>

As shown in Fig. 2S,† the simulated XRD pattern of the  $\alpha\text{-Fe}_2\text{O}_3$  bulk structure shows the strongest intensity of (104) orientation, which agrees with the experimental results.<sup>58</sup> However, based on our surface energy calculated results for the three dominant surface planes of hematite ( $\alpha\text{-Fe}_2\text{O}_3(104, 110, \text{ and } 012)$ ),  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface with the least surface energy is the most stable one, as shown in Table 1S,† Therefore, we selected the  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface.

The geometry optimization of  $\alpha\text{-Fe}_2\text{O}_3$  was carried out using a Monkhorst–Pack  $k$ -points scheme of  $5 \times 5 \times 2$  and  $2 \times 3 \times 1$  for bulk and surface calculations, respectively. For the final single-point runs for calculating the electronic properties such

as DOS, a mesh of  $6 \times 9 \times 1$  was used as shown in Fig. 1a. The  $\alpha\text{-Fe}_2\text{O}_3$  bulk structure contains iron and oxygen atoms arranged in a triangular hexagonal structure, and the calculated  $\alpha\text{-Fe}_2\text{O}_3$  bulk lattice parameters are ( $a = b = 5.0396 \text{ \AA}$  and  $c = 13.7494 \text{ \AA}$ ) in agreement with the experimental lattice parameter values ( $a = b = 5.036 \text{ \AA}$  and  $c = 13.75 \text{ \AA}$ ) and previous theoretical calculations.<sup>59</sup> The dopants calculated bulk lattice parameters in their bulk forms are ( $a = b = c = 2.953$ ,  $a = b = c = 4.003$ ,  $a = b = c = 3.689$ ,  $a = b = c = 4.194$ , and  $a = b = c = 4.076 \text{ \AA}$ ) for Fe, Ir, Cu, Co, and Ag, respectively, which is in agreement with the experimental lattice values.<sup>60–62</sup> The calculated chemical potential of the dopants is,  $-2.879$ ,  $-6.575$ ,  $-0.862$ ,  $-2.003$ , and  $-1.809$  for Fe, Ir, Cu, Co, and Ag, dopants, respectively, as shown in Table 2S,†

The calculated band gap of bulk  $\alpha\text{-Fe}_2\text{O}_3$  is 2.065 eV, as shown in Fig. 1b, which is within the experimental values (1.90–2.20 eV)<sup>63</sup> and similar to the previous theoretical study of 2.1 eV.<sup>64</sup>

The adsorption energy ( $E_{\text{ads}}$ ) of the methane on the surface was calculated as

$$E_{\text{ads}} = E_{\text{CH}_4+\text{surface}} - E_{\text{CH}_4} - E_{\text{surface}}$$

where  $E_{\text{CH}_4+\text{surface}}$  is the total energy of the catalyst on which methane is adsorbed,  $E_{\text{CH}_4}$  and  $E_{\text{surface}}$  are the energies of the  $\text{CH}_4$  molecule in the gas phase and the catalyst surface, respectively. We used the climbing-image nudged-elastic-band (CI-NEB)<sup>65</sup> method to evaluate the reaction energy barrier for methane activation, which locates the transition state (TS) structure. Vibrational frequency analyses were subsequently performed to verify the imaginary vibrational mode's uniqueness, confirming the saddle point's true nature. Charge redistributions during a surface reaction and the molecule's interaction with the surface were evaluated using the Bader charges method and partial density of states (PDOS) of atoms by considering before and after the adsorption of the  $\text{CH}_4$  molecule.

## 3. Results and discussion

### 3.1. Surface modeling

The  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface was modeled from the optimized bulk, consisting of nine atomic layers in which the bottom three atomic layers and seven atomic layers were fixed during the structural optimization and frequency calculations, respectively. A vacuum space of  $15 \text{ \AA}$  was used in the  $z$ -direction to avoid the interactions between the slabs. The  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface has three types of surface oxygen atoms: two-folded coordinated, three-folded coordinated on the top layer (denoted  $\text{O}_{2\text{C}}$  and  $\text{O}_{3\text{C}}$ , respectively), and four-folded coordinated in the subsurface (denoted as  $\text{O}_{4\text{C}}$ ). The corresponding minimum distances of three differently coordinated oxygen atoms with the nearest metal atom are  $1.79$ ,  $1.92$ , and  $2.12 \text{ \AA}$  for two-, three- and four-folded coordinated oxygen atoms, respectively. The most stable surface has been chosen based on its surface energy for methane activation, as shown in Table 1S,† The relative



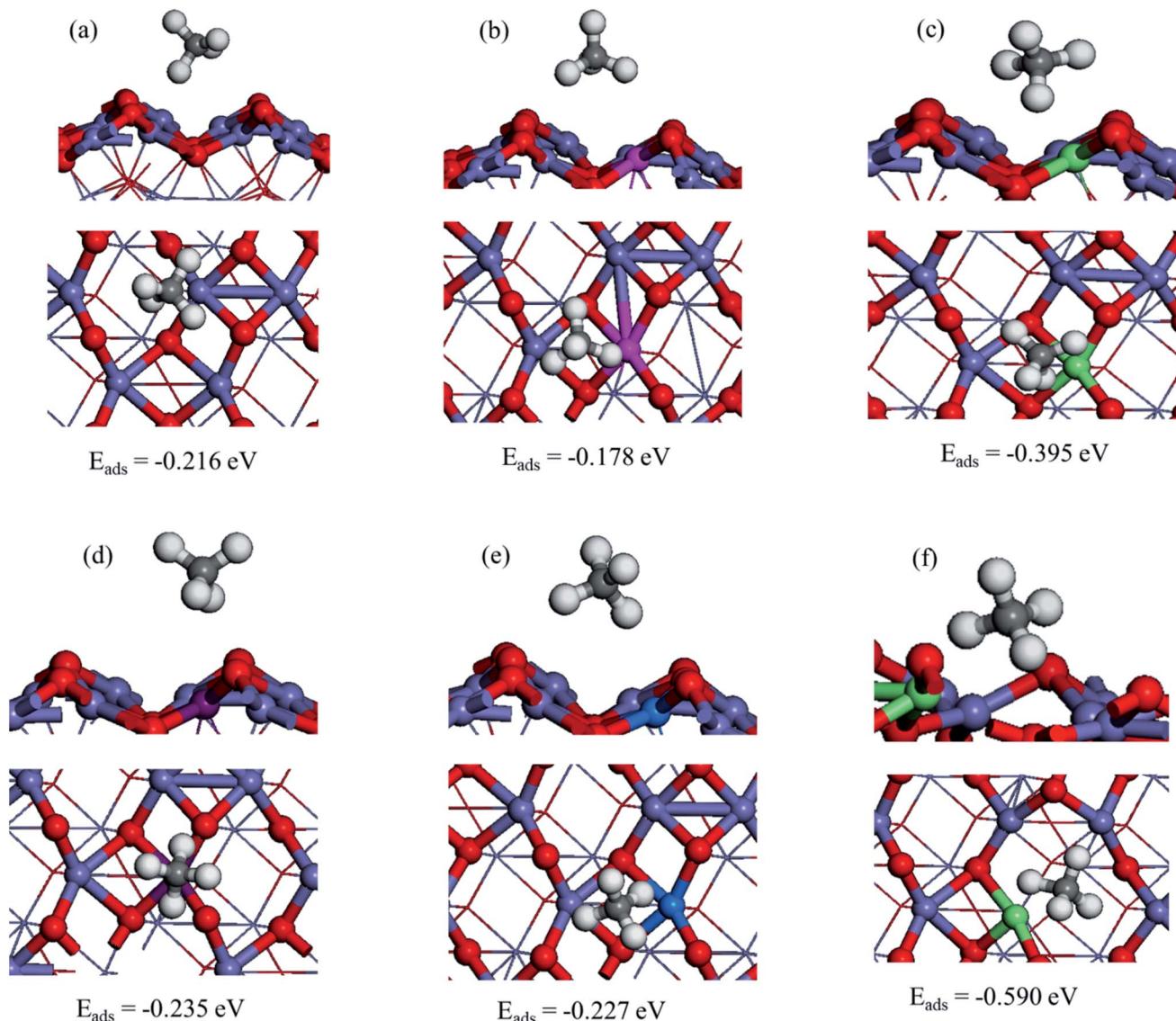


Fig. 2 Optimized structure of  $\text{CH}_4$  adsorption on  $\text{M}/\alpha\text{-Fe}_2\text{O}_3(110)$  surfaces: (a)  $\alpha\text{-Fe}_2\text{O}_3(110)$ , (b)  $\text{Ag}/\alpha\text{-Fe}_2\text{O}_3(110)$ , (c)  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , (d)  $\text{Co}/\alpha\text{-Fe}_2\text{O}_3(110)$ , (e)  $\text{Cu}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and (f)  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_v$  surfaces.

stability of  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface with different single metal dopants was compared by calculating formation energy as:

$$E_{\text{form}} = E_{\text{M}/\text{Fe}_2\text{O}_3} + \mu_{\text{Fe}} - E_{\text{Fe}_2\text{O}_3} - \mu_{\text{M}}$$

where  $E_{\text{M}/\text{Fe}_2\text{O}_3\text{M}}$  and  $E_{\text{M}/\text{Fe}_2\text{O}_3}$  are the total electronic energies of  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface after and before doping, respectively,  $\mu_{\text{Fe}}$  is the chemical potential of the iron atom replaced by single-atom dopants,  $\mu_{\text{M}}$  is the chemical potential of the dopants, and  $\text{M}$  is dopant atom (Ag, Ir, Cu, or Co). The chemical potential of the single-metal atom was assumed to be the electronic energy of a single atom in the bulk phase. Single metal atom doping site and the system geometry configuration is illustrated in Fig. 4S.†

### 3.2. Methane adsorption

To investigate the methane's most stable adsorption site over the pristine  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface, we considered Fe top, F-O

bridge,  $\text{O}_{2\text{C}}$  top, and  $\text{O}_{3\text{C}}$  top adsorption sites, as shown in Fig. 5S.† Our results indicate that  $\text{CH}_4$  adsorb preferentially on the Fe top of  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface with an adsorption energy of  $-0.211$  eV by coordinating one of the C-H bonds toward the surface shown in Fig. 2a. The adsorption is relatively strong compared to those of the  $\alpha\text{-Fe}_2\text{O}_3(0001)$  surface, on which the adsorption energy is  $-0.1$  eV. The  $\text{C}\cdots\text{Fe}$  and  $\text{CH}\cdots\text{O}$  distances are  $2.719$  and  $2.480$  Å, respectively; one of the C-H bond lengths of  $\text{CH}_4$  extended to  $1.109$  Å from  $1.097$  Å (free molecule) upon the adsorption.<sup>66</sup>

Further, we obtained the most stable adsorption configuration of  $\text{CH}_4$  on different metal-doped hematite surfaces ( $\text{M}/\alpha\text{-Fe}_2\text{O}_3(110)$ ) where ( $\text{M} = \text{Ag, Cu, Ir, or Co}$ ). The optimized most stable  $\text{CH}_4$  adsorption configurations on  $\text{M}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface are shown in Fig. 2, 6S and 7S of ESI† showing the other less stable structures. The calculated adsorption energies, selective bond lengths, and distances of  $\text{CH}_4$  adsorption on  $\text{M}/$



**Table 1** Calculated adsorption energies ( $E_{\text{ads}}$  in eV), C–H bond lengths ( $d_{\text{C–H}}$  in Å), H to surface oxygen distance ( $d_{\text{CH}\cdots\text{O}}$  in Å), C to nearest metal atom distances ( $d_{\text{C}\cdots\text{M}}$  in Å), activation barriers ( $E_a$  in eV), and reaction energies ( $\Delta E$  in eV) for  $\text{CH}_4$  activation on different catalyst surfaces

Surface	$E_{\text{ads}}$	$d_{\text{C–H}}$	$d_{\text{CH}\cdots\text{O}}$	$d_{\text{C}\cdots\text{M}}$	$E_a$	$\Delta E$	Reference
Free $\text{CH}_4$ molecule	—	1.097	—	—	—	—	This study
$\alpha\text{-Fe}_2\text{O}_3(110)$	−0.216	1.109	2.480	2.719	0.819	0.358	This study
$\text{Ag}/\alpha\text{-Fe}_2\text{O}_3(110)$	−0.178	1.100	2.729	3.398	—	—	This study
$\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$	−0.395	1.146	2.364	1.918	0.439	0.300	This study
$\text{Co}/\alpha\text{-Fe}_2\text{O}_3(110)$	−0.235	1.099	2.701	3.797	—	—	This study
$\text{Cu}/\alpha\text{-Fe}_2\text{O}_3(110)$	−0.227	1.100	3.018	3.377	—	—	This study
$\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$	−0.590	1.182	2.805	1.794	0.585	−0.590	This study
$\text{IrO}_2(110)$	−0.41	1.16	2.11	—	0.300	−1.090	19
$\text{Pd}/\text{CeO}_2(111)$	−0.19	—	—	—	1.140	—	35
$\text{Pd}/\text{Cu}(111)$	−0.20	—	—	—	1.460	0.940	28
$\text{IrO}_2/\text{TiO}_2(110)$	−0.74	1.15	2.24	—	0.290	−1.080	68

$\alpha\text{-Fe}_2\text{O}_3(110)$  surfaces (M = Ag, Ir, Co, or Cu) are listed in Table 1. The  $\text{CH}_4$  on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface has the largest adsorption energy and most extended C–H bond length compared to the other considered single-metal doped systems and the pristine surface. The C–H bond length is elongated to 1.146 Å from 1.097 Å (free molecule), indicating that the C–H has been significantly activated. Furthermore, the  $\text{CH}\cdots\text{O}$  and  $\text{C}\cdots\text{Ir}$  distances (2.364 and 1.918 Å, respectively) are much shorter than those single-metal doped systems and the pristine surface due to the more vital interaction between  $\text{CH}_4$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface.

Among the considered different single metal-doped  $\alpha\text{-Fe}_2\text{O}_3(110)$  surfaces, Ir single atom doped  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface ( $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ ) has the  $\text{CH}_4$  adsorption energy of −0.395 eV. As shown in Table 1, M/ $\alpha\text{-Fe}_2\text{O}_3(110)$  surface (M = Ir) has the best results in C–H bond elongation, interaction, and  $\text{CH}_4$  adsorption energy. The C–H bond elongation on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface has a similar value to with  $\text{IrO}_2(110)$  surface catalyst, *i.e.*, 1.15 Å. Researchers agreed that  $\text{IrO}_2(110)$  surface catalyst has a remarkable performance for  $\text{CH}_4$  activation.<sup>67,68</sup> Hence  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface catalyst has nearly similar activity to the  $\text{IrO}_2(110)$  surface in terms of C–H bond elongation. As a result, this catalyst will be one of the promising and economically feasible catalysts for  $\text{CH}_4$  activation. Therefore, we selected the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface catalyst for further study.

As the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface exhibits enhanced catalytic activity toward  $\text{CH}_4$  adsorption among the others, we further considered the presence of single oxygen vacancy on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface ( $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$ ) and investigated the reactivity towards  $\text{CH}_4$  activation. The oxygen vacancy formation energy  $E_\text{O}_\text{V}$  was calculated using the following equation:

$$E_\text{O}_\text{V} = E_{\text{Ir}/\text{Fe}_2\text{O}_3+\text{O}_\text{V}} + \frac{1}{2}E_{\text{O}_2} - E_{\text{sur}}$$

where  $E_{\text{Ir}/\text{Fe}_2\text{O}_3+\text{O}_\text{V}}$  is the total energy of the surface with a single oxygen vacancy,  $E_{\text{sur}}$  is the electronic energy of the pure surface, and  $E_{\text{O}_2}$  is the total energy of the oxygen molecule in the gas phase. We calculated the  $E_\text{O}_\text{V}$  formation energy from the M/ $\alpha\text{-Fe}_2\text{O}_3(110)$  surfaces by pulling out one oxygen atom from the top surface of the structure. From our study, the  $E_\text{O}_\text{V}$  formation energies of the twofold coordinated and threefold oxygen atoms

from the top layer of the pristine  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface were 2.76 and 2.93 eV, respectively, similar to the previous study.<sup>39</sup> Similarly,  $E_\text{O}_\text{V}$  formation energy of the twofold coordinated and threefold coordinated oxygen atoms from the top layer of the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface was considered. However, the structure became unstable when we created the twofold coordinated oxygen atom defect from  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface. The  $E_\text{O}_\text{V}$  formation energy of the threefold coordinated oxygen atoms from the top layer of the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface was 2.55 eV. As a result, we used threefold coordinated oxygen vacancies to investigate the effect of oxygen vacancies on  $\text{CH}_4$  adsorption; a similar method was applied in a previous study.<sup>69</sup>

In the same case, different adsorption sites were considered on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface, and each selected configuration was optimized to find a stable structure. The calculated adsorption energy for the different adsorption sites and selected geometry parameters are given in Fig. 8S.† As it has been seen in Fig. 2f, it is the most stable  $\text{CH}_4$  adsorption site on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface with the adsorption energy of −0.590 eV, which is much larger than that of other surfaces considered in this study. In the process of adsorption, the elongation of the activated C–H bond of  $\text{CH}_4$  on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface (1.182 Å) was the greatest compared to that on other surfaces. Recent study<sup>39</sup> also indicates that oxygen vacancy enhanced the C–H bond activation in doped metal oxides.  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface increased the C–H bond length by 0.085 Å compared to the gaseous methane C–H bond length. The  $\text{CH}\cdots\text{O}$  and  $\text{C}\cdots\text{Ir}$  distances on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface are 2.805 and 1.794 Å, respectively. Compared to  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface, the  $\text{CH}\cdots\text{O}$  distance becomes longer, whereas the  $\text{C}\cdots\text{Ir}$  distance is shorter, indicating that the agostic interaction increases, whereas the  $\text{CH}\cdots\text{O}$  interaction decreases in the case of  $\text{CH}_4$  adsorption on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surface. The methane molecule's C–H bond elongation on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surfaces was due to the  $\text{CH}\cdots\text{O}$  and agostic interactions between C–H of  $\text{CH}_4$  and surfaces.

### 3.3. Electronic structure analysis

For a detailed comparison of the electronic behavior of  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surfaces upon  $\text{CH}_4$  adsorption; we plotted EDD and the DOS distributions for the most



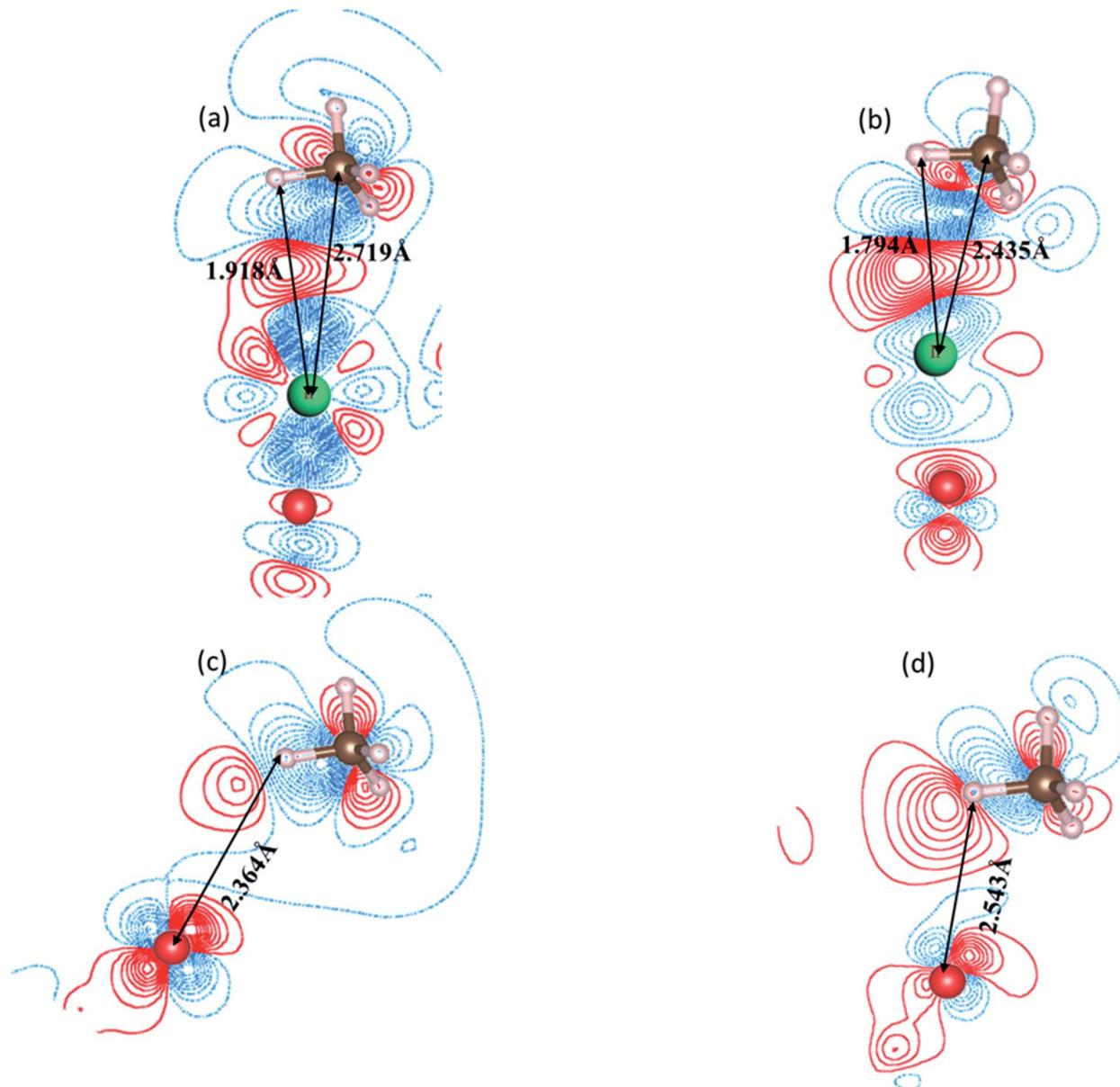


Fig. 3 2-D plots of the calculated electron density difference (EDD) for the  $\text{CH}_4$  adsorption. (a)  $\text{C}-\text{H}\cdots\text{Ir}$  coplanar on  $\text{Ir}/\text{Fe}_2\text{O}_3(110)$  surface, (b)  $\text{C}-\text{H}\cdots\text{Ir}$  coplanar on  $\text{Ir}/\text{Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surface, (c)  $\text{C}-\text{H}\cdots\text{O}$  coplanar on  $\text{Ir}/\text{Fe}_2\text{O}_3(110)$  surface and (d)  $\text{C}-\text{H}\cdots\text{O}$  coplanar on  $\text{Ir}/\text{Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surface. The red and blue lines represent the increasing and decreasing electron densities, respectively. The isovalue is  $0.002 \text{ ebohr}^{-3}$ .

stable adsorption configurations. Fig. 3 shows the EDD contours of  $\text{CH}_4$  adsorption on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surfaces with two different cutting planes ( $\text{C}-\text{H}\cdots\text{Ir}$  and  $\text{C}-\text{H}\cdots\text{O}_\text{cus}$ ). As shown in Fig. 3a and b, electron accumulation between one  $\text{C}-\text{H}$  bond of methane and Ir atom on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surfaces were observed, which indicates the existence of agostic interaction between methane and surface. Further, as shown in Fig. 3c and d, it could be observed the electron accumulation between one of the  $\text{H}(\text{C})$  and surface oxygen atoms on both  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surfaces, indicating the presence of  $\text{CH}\cdots\text{O}$  hydrogen bond between methane and the surface. The agostic interaction will make the electron transfer from

methane to the surface, whereas  $\text{CH}\cdots\text{O}$  hydrogen bonding causes the electron transfer from the surface to methane. Therefore, these two cooperative interactions between methane and the surface result in great methane adsorption energy on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface.<sup>19,71</sup> The strong interaction between methane and surface results in significant  $\text{C}-\text{H}$  bond elongation, indicating that  $\text{CH}_4$  has significantly been activated on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_\text{v}$  surfaces, as illustrated in Fig. 3. To understand the effect of the van der Waals force on methane adsorption, we calculated the methane adsorption energy without considering the van der Waals force correction. As shown in Table 3S,<sup>†</sup> on the surfaces of  $\alpha\text{-Fe}_2\text{O}_3(110)$ ,  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)-\text{O}_\text{v}$ , the



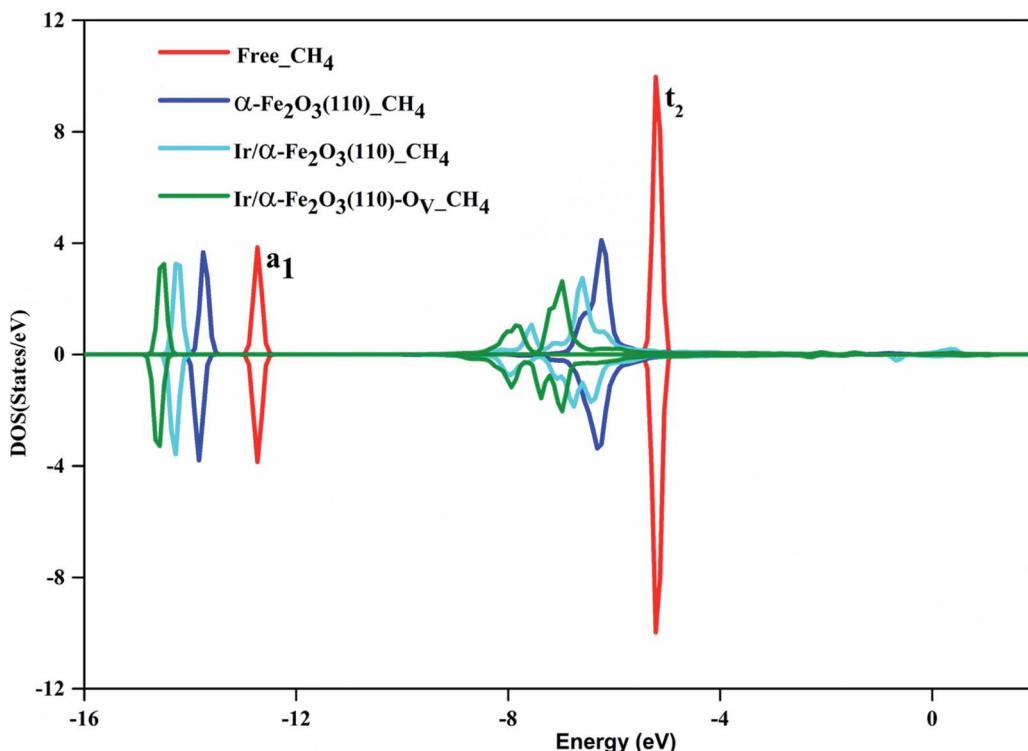


Fig. 4 The density of states of  $\text{CH}_4$ : free molecule (red line), on  $\alpha\text{-Fe}_2\text{O}_3(110)$  (blue line),  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surfaces (cyan line), and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surfaces (green line), respectively. The upper and lower panels represent spin up and spin down, respectively.

contribution of van der Waals force to methane adsorption is  $-0.17$ ,  $-0.22$ ,  $-0.32$  eV, respectively. It is like the results of the literature.<sup>70,71</sup>

Fig. 4 shows the DOS of the  $\text{CH}_4$  molecule before and after adsorbed on  $\alpha\text{-Fe}_2\text{O}_3(110)$ ,  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surfaces. The DOS plot of the free  $\text{CH}_4$  molecule shows there are two peaks located at around  $-5.1$  and  $-12.6$  eV, which belong to the  $t_2$  and  $a_1$  states of  $\text{CH}_4$   $\sigma$  hybridization states, respectively. After adsorption, both states shift to lower energy regions, and the  $t_2$  state becomes much broad. The order of amplitude of shift and broadening is  $\alpha\text{-Fe}_2\text{O}_3(110) > \text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110) > \text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$ , related to adsorption energy. The interaction of  $\text{CH}_4$  with  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  is the strongest, resulting in the most extensive degree of state energy downshift and broadening. The Bader charge calculation found that  $0.034$ ,  $0.143$ , and  $0.145$  net electrons/charges were transferred from the  $\text{CH}_4$  molecule to  $\alpha\text{-Fe}_2\text{O}_3(110)$ ,  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surfaces, respectively. The last two have large electrons transfer from methane to the surface, indicating that the agostic interactions are large, which causes the great C–H activation.

### 3.4. Methane activation

The activation of the first C–H bond is the critical and the rate-limiting step in methane conversion on most catalysts. In this work, we predominantly calculated the first dehydrogenation of

$\text{CH}_4$  over  $\alpha\text{-Fe}_2\text{O}_3(110)$ ,  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surfaces. The calculated energy barrier and reaction energy profile for  $\text{CH}_4$  activation over these three catalysts are illustrated in Fig. 5, and the structure of the initial state ( $\text{CH}_4^*$ ), transition state (TS), and intermediate state (dis- $\text{CH}_4$ ) are given in Fig. 6. As shown in Fig. 5, the dehydrogenation barrier of the  $\text{CH}_4$  molecule on the  $\alpha\text{-Fe}_2\text{O}_3(110)$  surface is the largest with  $0.819$  eV, and the reaction energy is endothermic by  $0.358$  eV. The dehydrogenation barrier of the  $\text{CH}_4$  molecule on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface is lowest at  $0.439$  eV, and reaction energy is slightly endothermic at  $0.297$  eV. However, the adsorption energy is  $-0.395$  eV, smaller than the reaction barrier. Although the C–H activation barrier on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surface is  $0.585$  eV, which is slightly higher than that on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  surface, it has larger desorption energy making the  $\text{CH}_4$  activation reaction on the  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surface can occur through a mediated mechanism and at a lower temperature. In addition, the  $\text{CH}_4$  reaction on  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surface is exothermic by  $0.585$  eV. Though the  $\text{CH}_4$  adsorption energy and its C–H activation barrier over  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_v$  surface compete with each other, the activation of the adsorbent on the catalyst could be proceeded by adapting the suitable reaction conditions or introducing Lewis–Brønsted acid site pairs in the catalyst.<sup>72,73</sup> Lercher *et al.* recently proposed the Lewis–Brønsted acid mechanism, in which the Lewis–Brønsted acid site pair dehydrogenates alkanes *via* a bifunctional mechanism that is more active than the isolated Lewis and Brønsted acid sites.<sup>73</sup>



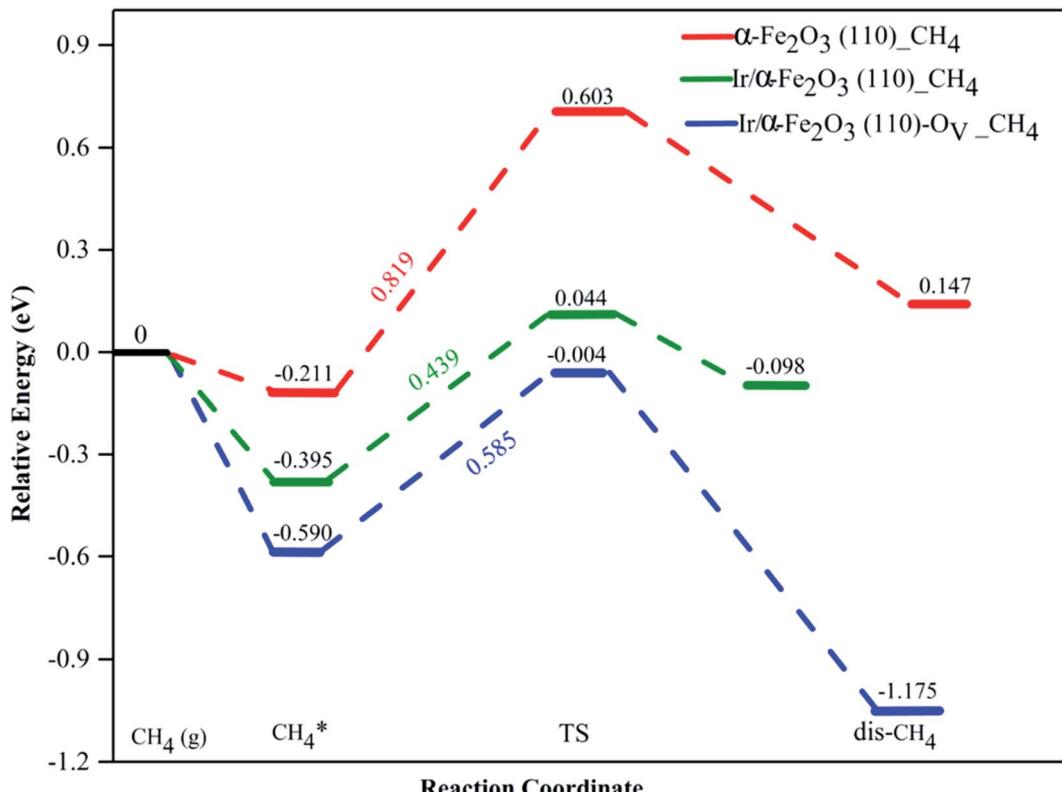


Fig. 5 Potential energy profile of  $\text{CH}_4$  activation on  $\alpha\text{-Fe}_2\text{O}_3(110)$  (red dash line),  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$  (green dash line), and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surfaces (blue dash line).

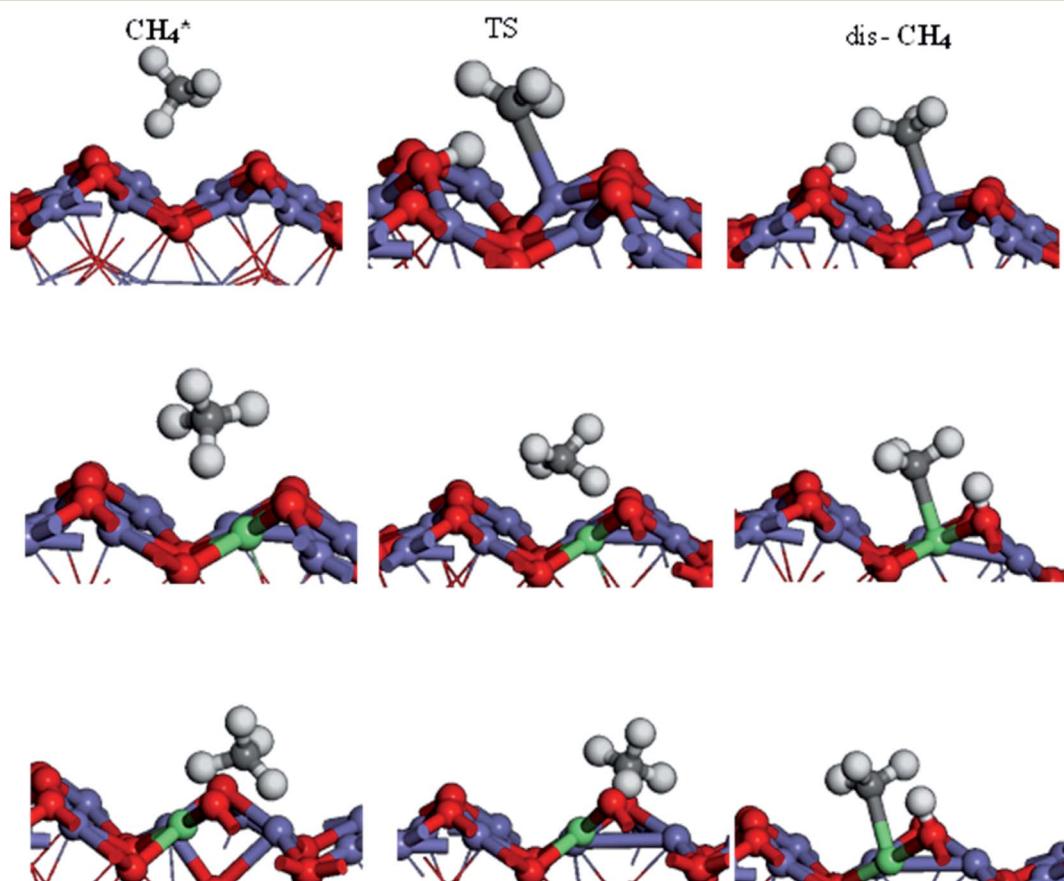


Fig. 6 Side view of initial, transition, and intermediate states of  $\text{CH}_4$  oxidation on  $\alpha\text{-Fe}_2\text{O}_3(110)$ ,  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)$ , and  $\text{Ir}/\alpha\text{-Fe}_2\text{O}_3(110)\text{-O}_\text{V}$  surfaces.

The first C–H bond activation on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is higher than pure IrO<sub>2</sub>(110), and IrO<sub>2</sub>/TiO<sub>2</sub>(110) surfaces but much smaller compared to Pd/Cu(111) and Pd/CeO<sub>2</sub>(111) surfaces, as shown in Table 1. After activation, the C–H bond length of the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is the longest, but the activation energy is higher than that of the pristine IrO<sub>2</sub>(110) and IrO<sub>2</sub>/TiO<sub>2</sub>(110) surfaces. It may be due to the weaker CH···O hydrogen bonding since the CH···O distance on the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is longer than on the other surfaces. Likewise, the reaction energy for CH<sub>4</sub> activation on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is slightly less exothermic compared on IrO<sub>2</sub>(110) and IrO<sub>2</sub>/TiO<sub>2</sub>(110) surfaces but much exothermic than on Pd/Cu(111) surface. Although compared with IrO<sub>2</sub>(110) and IrO<sub>2</sub>/TiO<sub>2</sub>(110) surfaces, Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is slightly less favorable thermodynamically and kinetically for CH<sub>4</sub> activation. However, this catalyst uses a relatively small amount of expensive Ir, and the activation energy is still quite low, making it a viable candidate for CH<sub>4</sub> activation both thermodynamically and kinetically.

To examine the relationship between activation barriers ( $E_a$ ) and adsorption energies ( $E_{ads}$ ) for C–H dissociation of methane on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), and Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surfaces, we found that they have no linear relationship, as shown in Fig. 3S.† The bond lengths for C–H activation on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110), and Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surfaces are 1.408, 1.312, and 1.147 Å, respectively. It indicates that the C–H activation exists in late transition on the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface, and early transition state on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface, resulting in no linear relationship between adsorption energy and activation barrier.

Previous studies have shown that IrO<sub>2</sub> has excellent catalytic activity for methane dehydrogenation.<sup>74,75</sup> But methane is completely oxidized to CO<sub>2</sub> and H<sub>2</sub>O.<sup>76,77</sup> In this study, we only considered a single metal doping and a single oxygen vacancy which is very active for methane adsorption. Cheng *et al.* studied the adsorption and dissociation of methane on ferric oxide oxygen carriers and they found that the activation barrier of C–H decreased when the surface oxygen vacancy concentration was increased from 0%–2.67%.<sup>78</sup> In the future, we will further investigate whether increasing the concentration of dopants and oxygen vacancies can further improve catalytic activity.

## 4. Conclusions

In this study, the DFT + *U* method calculations were performed to study the C–H bond activation of CH<sub>4</sub> molecules on the surfaces of pristine  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) and doped  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) with single-atom (Ag, Ir, Co, or Cu). From calculation results, we found that the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface has considerable interaction with CH<sub>4</sub> molecules compared to other considered single-atom doped surfaces. Furthermore, we investigated the activation of CH<sub>4</sub> on the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface with oxygen vacancy. EDD, DOS, and Bader charge analyses show that the co-existence of CH···O hydrogen bonding and agostic interactions between methane and surfaces results in large adsorption energies of –0.395 and –0.590 eV on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) and Ir/ $\alpha$ -

Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surfaces, respectively, and significantly elongated the length of one C–H bond on methane. Although the C–H activation barrier on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is slightly higher than that on Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110) surface, it has larger desorption energy making the CH<sub>4</sub> activation reaction on the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface can occur through a mediated mechanism and at a lower temperature. The calculation indicates that the first dehydrogenation of the CH<sub>4</sub> molecule on the Ir/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(110)–O<sub>V</sub> surface is thermodynamically and kinetically favorable.

## Conflicts of interest

The authors declare that there is no conflict of interest.

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## References

- 1 P. Tang, Q. J. Zhu, Z. X. Wu and D. Ma, *Energy Environ. Sci.*, 2014, **7**, 2580–2591.
- 2 P. Schwach, X. L. Pan and X. H. Bao, *Chem. Rev.*, 2017, **117**, 8497–8520.
- 3 E. McFarland, *Science*, 2012, **338**, 340–342.
- 4 R. A. Kerr, *Science*, 2010, **328**, 1624–1626.
- 5 E. D. Sloan, *Nature*, 2003, **426**, 353–359.
- 6 N. N. Xu, C. A. Coco, Y. D. Wang, T. S. Su, Y. Wang, L. W. Peng, Y. X. Zhang, Y. Y. Liu, J. L. Qiao and X. D. Zhou, *Appl. Catal., B*, 2021, **282**, 119572.
- 7 J. H. Lunsford, *Catal. Today*, 2000, **63**, 165–174.
- 8 W. Nabgan, T. A. T. Abdullah, R. Mat, B. Nabgan, Y. Gambo, M. Ibrahim, A. Ahmad, A. A. Jalil, S. Triwahyono and I. Saeh, *Renewable Sustainable Energy Rev.*, 2017, **79**, 347–357.
- 9 T. T. R. Karl and K. E. Trenberth, *Science*, 2003, **302**, 1719–1723.
- 10 D. Hu, V. V. Ordonsky and A. Y. Khodakov, *Appl. Catal., B*, 2021, **286**, 119913.
- 11 X. G. Guo, G. Z. Fang, G. Li, H. Ma, H. J. Fan, L. Yu, C. Ma, X. Wu, D. H. Deng, M. M. Wei, D. L. Tan, R. Si, S. Zhang, J. Q. Li, L. T. Sun, Z. C. Tang, X. L. Pan and X. H. Bao, *Science*, 2014, **344**, 616–619.
- 12 Y. L. Wang, P. Hu, J. Yang, Y. A. Zhu and C. Chen, *Chem. Soc. Rev.*, 2021, **50**, 4299–4358.
- 13 S. Q. Wu, L. Z. Wang and J. L. Zhang, *J. Photochem. Photobiol., C*, 2021, **46**, 100400.
- 14 X. Meng, X. Cui, N. P. Rajan, L. Yu, D. Deng and X. Bao, *Chem*, 2019, **5**, 2296–2325.
- 15 P. Tomkins, M. Ranocchiai and J. A. van Bokhoven, *Acc. Chem. Res.*, 2017, **50**, 418–425.
- 16 S. K. Li, R. Ahmed, Y. H. Yi and A. Bogaerts, *Catalysts*, 2021, **11**, 590.



17 T. L. M. Pham, E. G. Leggesse and J. C. Jiang, *Catal. Sci. Technol.*, 2015, **5**, 4064–4071.

18 T. L. M. Pham, S. Nachimuthu, J. L. Kuo and J. C. Jiang, *Appl. Catal., A*, 2017, **541**, 8–14.

19 C.-C. Wang, S. S. Siao and J.-C. Jiang, *J. Phys. Chem. C*, 2012, **116**, 6367–6370.

20 Z. Liang, T. Li, M. Kim, A. Asthagiri and J. F. Weaver, *Science*, 2017, **356**, 298–301.

21 R. Martin, M. Kim, A. Franklin, Y. X. Bian, A. Asthagiri and J. F. Weaver, *Phys. Chem. Chem. Phys.*, 2018, **20**, 29264–29273.

22 Y. X. Bian, M. Kim, T. Li, A. Asthagiri and J. F. Weaver, *J. Am. Chem. Soc.*, 2018, **140**, 2665–2672.

23 C. H. Yeh, T. M. L. Pham, S. Nachimuthu and J. C. Jiang, *ACS Catal.*, 2019, **9**, 8230–8242.

24 O. Ola and M. M. Maroto-Valer, *Appl. Catal., A*, 2015, **502**, 114–121.

25 P. D. Murzin, A. A. Murashkina, A. V. Emeline and D. W. Bahnemann, *Top. Catal.*, 2020, **64**, 817–823.

26 B. Rusinque, S. Escobedo and H. de Lasa, *Catalysts*, 2020, **10**, 74.

27 H. R. Eisenberg and R. Baer, *J. Phys. Chem. C*, 2015, **119**, 196–215.

28 Y. Meng, C. Ding, X. Gao, K. Zhang, J. Wang and Z. Li, *Appl. Surf. Sci.*, 2020, **513**, 145724.

29 F. Guo, J. Y. Ran, J. T. Niu, H. Y. Qiu and Z. L. Ou, *Int. J. Energy Res.*, 2021, **45**, 10100–10111.

30 Z. Feng, X. Liu, Y. Wang and C. Meng, *Molecules*, 2021, **26**, 2234.

31 Z. Xu, Y. Yue, X. Bao, Z. Xie and H. Zhu, *ACS Catal.*, 2019, **10**, 818–828.

32 L. Xie, R. Wang, Y. Chai, X. Weng, N. Guan and L. Li, *J. Energy Chem.*, 2021, **63**, 262–269.

33 B. Li and H. Metiu, *J. Phys. Chem. C*, 2011, **115**, 18239–18246.

34 A. D. Mayernick and M. J. Janik, *J. Phys. Chem. C*, 2008, **112**, 14955–14964.

35 Y.-Q. Su, I. A. Filot, J.-X. Liu and E. J. Hensen, *ACS Catal.*, 2018, **8**, 75–80.

36 O. Akhavan, *Appl. Surf. Sci.*, 2010, **257**, 1724–1728.

37 F. Shi, M. K. Tse, M.-M. Pohl, A. Brückner, S. Zhang and M. Beller, *Angew. Chem., Int. Ed.*, 2007, **119**, 9022–9024.

38 J. Liang, Q. H. Zhang, H. L. Wu, G. Y. Meng, Q. H. Tang and Y. Wang, *Catal. Commun.*, 2004, **5**, 665–669.

39 N. J. O'Connor, A. Jonayat, M. J. Janik and T. P. Senftle, *Nat. Catal.*, 2018, **1**, 531–539.

40 G. Kresse and J. Hafner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **47**, 558.

41 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.

42 J. Klimeš, D. R. Bowler and A. Michaelides, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **83**, 195131.

43 J. Klimeš, D. R. Bowler and A. Michaelides, *J. Phys.: Condens. Matter*, 2009, **22**, 022201.

44 J. Klimeš and A. Michaelides, *J. Chem. Phys.*, 2012, **137**, 120901.

45 G. Kresse and D. Joubert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758–1775.

46 M. Arroyo-de Dompablo, A. Morales-García and M. Taravillo, *J. Chem. Phys.*, 2011, **135**, 054503.

47 C. W. Castleton, A. Lee and J. Kullgren, *J. Phys. Chem. C*, 2019, **123**, 5164–5175.

48 Y. Ping, G. Galli and W. A. Goddard III, *J. Phys. Chem. C*, 2015, **119**, 11570–11577.

49 C. Zhao, B. Li, X. Zhou, J. Chen and H. Tang, *Metals*, 2021, **11**, 424.

50 S. Sagadevan, K. Pal and Z. Z. Chowdhury, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 12591–12597.

51 Y. Wang, H. Ge, Y. Chen, X. Meng, J. Ghanbaja, D. Horwat and J. Pierson, *Chem. Commun.*, 2018, **54**, 13949–13952.

52 L.-H. Tjeng, M. B. Meinders, J. van Elp, J. Ghijssen, G. A. Sawatzky and R. L. Johnson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1990, **41**, 3190.

53 V. Tripkovic, H. A. Hansen and T. Vegge, *ACS Catal.*, 2017, **7**, 8558–8571.

54 Y. Tang, S. Zhao, B. Long, J.-C. Liu and J. Li, *J. Phys. Chem. C*, 2016, **120**, 17514–17526.

55 W. Li, S. G. Srinivasan, D. Salahub and T. Heine, *Phys. Chem. Chem. Phys.*, 2016, **18**, 11139–11149.

56 M. Jamal, S. S. Nishat and A. Sharif, *Chem. Phys.*, 2021, **545**, 111160.

57 M. Méndez-Galván, C. A. Celaya, O. A. Jaramillo-Quintero, J. Muniz, G. Díaz and H. A. Lara-García, *Nanoscale Adv.*, 2021, **3**, 1382–1391.

58 M. Khalil, J. J. Yu, N. Liu and R. L. Lee, *J. Nanopart. Res.*, 2014, **16**, 183–187.

59 L. Qin, Z. Cheng, M. Q. Guo, M. Y. Xu, J. A. Fan and L. S. Fan, *ACS Energy Lett.*, 2017, **2**, 70–74.

60 P. Haas, F. Tran and P. Blaha, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **79**, 085104.

61 H. Eguchi, K. Tsumuraya, T. Nagano and S. Kihara, *Mater. Trans., JIM*, 1999, **40**, 1198–1204.

62 A. I. o. Physics, and D. E. Gray, *American Institute of Physics Handbook*, McGraw-Hill, 3rd edn, 1972.

63 F. Kraushofer, Z. Jakub, M. Bichler, J. Hulva, P. Drmota, M. Weinold, M. Schmid, M. Setvin, U. Diebold and P. Blaha, *J. Phys. Chem. C*, 2018, **122**, 1657–1669.

64 N. Y. Dzade, A. Roldan and N. H. de Leeuw, *Minerals*, 2014, **4**, 89–115.

65 G. Henkelman, B. P. Uberuaga and H. Jonsson, *J. Chem. Phys.*, 2000, **113**, 9901–9904.

66 J.-J. Tang and B. Liu, *J. Phys. Chem. C*, 2016, **120**, 6642–6650.

67 Y. Tsuji and K. Yoshizawa, *J. Phys. Chem. C*, 2018, **122**, 15359–15381.

68 S. Nachimuthu, H.-J. Lai, Y.-C. Chen and J.-C. Jiang, *Appl. Surf. Sci.*, 2022, **577**, 151938.

69 Y. Zhao, K. R. Yang, Z. Wang, X. Yan, S. Cao, Y. Ye, Q. Dong, X. Zhang, J. E. Thorne and L. Jin, *Proc. Natl. Acad. Sci. U. S. A.*, 2018, **115**, 2902–2907.

70 S. González, F. Viñes, J. F. García, Y. Erazo and F. Illas, *Surf. Sci.*, 2014, **625**, 64–68.

71 M. J. Tillotson, P. M. Brett, R. A. Bennett and R. Grau-Crespo, *Surf. Sci.*, 2015, **632**, 142–153.



72 X. Liu, M. Xu, L. Wan, H. Zhu, K. Yao, R. Linguerri, G. Chambaud, Y. Han and C. Meng, *ACS Catal.*, 2020, **10**, 3084–3093.

73 M. W. Schreiber, C. P. Plaisance, M. Baumgärtl, K. Reuter, A. Jentys, R. Bermejo-Deval and J. A. Lercher, *J. Am. Chem. Soc.*, 2018, **140**, 4849–4859.

74 Y. Tsuji and K. Yoshizawa, *J. Phys. Chem. C*, 2020, **124**, 17058–17072.

75 J. Xu, X.-M. Cao and P. Hu, *J. Phys. Chem. C*, 2019, **123**, 28802–28810.

76 M. Kim, A. Franklin, R. Martin, F. Feng, T. Li, Z. Liang, A. Asthagiri and J. Weaver, *J. Phys. Chem. C*, 2019, **123**, 27603–27614.

77 R. Martin, C. J. Lee, V. Mehar, M. Kim, A. Asthagiri and J. F. Weaver, *ACS Catal.*, 2022, **12**, 2840–2853.

78 Z. Cheng, L. Qin, M. Guo, J. A. Fan, D. Xu and L.-S. Fan, *Phys. Chem. Chem. Phys.*, 2016, **18**, 16423–16435.

