

REVIEW

[View Article Online](#)
[View Journal](#) | [View Issue](#)



Cite this: *Mater. Chem. Front.*, 2017, 1, 1709

Received 9th January 2017,
Accepted 19th February 2017

DOI: 10.1039/c7qm00007c

rsc.li/frontiers-materials

Metal organic frameworks as precursors for the manufacture of advanced catalytic materials

Lide Oar-Arteta, Tim Wezendonk, Xiaohui Sun, Freek Kapteijn and Jorge Gascon*

The use of metal organic frameworks as hard templates for the preparation of heterogeneous catalysts is thoroughly reviewed. In this critical article, the main factors to consider when using a MOF as a sacrificial template are first discussed. Then, the existing literature on the topic is reviewed, classifying the different examples according to the MOF metal. Finally, the main advantages, limitations and perspectives of the so-called MOF mediated synthesis are outlined.

1. Introduction

Translating material design into an actual catalyst through controlled synthesis is one of the most important challenges heterogeneous catalysis faces nowadays. Despite many advances, conventional synthetic techniques usually fall short in delivering materials with the desired properties. The preparation of metal nanoparticle based catalysts, where it is not always possible to achieve desired metal loadings and controlled particle size, is an outstanding example of this challenge. Recent advances

Catalysis Engineering, Department of Chemical Engineering, Delft University of Technology, Van der Maasweg 9, 2629 HZ Delft, The Netherlands.
E-mail: j.gascon@tudelft.nl



Lide Oar-Arteta

interests focus on the preparation of metal-organic framework derived catalysts for their application in different catalytic processes and with special emphasis on the Fischer-Tropsch synthesis.

Lide Oar-Arteta (1986) received her PhD (2014) from the University of The Basque Country (UPV/EHU) after a secondment in the Institut de Chimie des Milieux et Matériaux de Poitiers (IC2MP) in 2012. In 2015, she joined the Catalysis Engineering group of Delft University of Technology (The Netherlands) where she is now a Marie-Slodowska Curie postdoctoral researcher in collaboration with Haldor Topsøe A/S (Denmark). Her current research

in this field include (i) the use of colloidal,¹ reverse micelle,² and dendrimer chemistry³ in the production of metal and metal oxide nanoparticles with well-defined sizes, shapes, and compositions, (ii) the introduction of sol-gel⁴ and atomic layer deposition⁵ chemistry for the production and modification of high-surface-area supports and active phases, (iii) the design of dumbbells, core@shell, and other complex nanostructures which lead to multiple functionality in catalysis⁶ and (iv) the mixed molecular-nanostructure approach that can be used to develop more demanding catalytic sites, by derivatization of the surface of solids or tethering or immobilizing other chemical functionalities.⁷

Metal organic frameworks (MOFs) are crystalline solids consisting of infinite lattices built up from inorganic secondary



Tim A. Wezendonk

investigating the structural and elemental influence of various MOFs, and elucidating the evolution of the active phase of MOF-derived Fe catalysts during Fischer-Tropsch operation.



building units (SBUs, metal ions or clusters) and organic linkers, connected by coordination bonds of moderate strength. MOFs are among the materials with the highest porosity, and their pore size, shape, dimensionality, and chemical environment can be finely tuned.⁸ As a consequence, MOFs have been extensively studied not only in catalysis but also in different fields such as fuel storage, batteries, and supercapacitors.⁹ Specially in catalysis, the high versatility of MOF design provides clear advantages, since in principle, it should be possible to rationally design not only the active site but also its environment with an unprecedented degree of precision.¹⁰ However, the main constraint for MOF application in catalysis is their relatively low thermal and chemical stability, especially when compared with inorganic porous solids. Although a handful of these materials are particularly robust,^{11–14} it is very difficult to anticipate the stability of a given MOF under certain reaction conditions, even when the thermal stability is very high.



Xiaohui Sun

Xiaohui Sun received his BS degree (2010) and MS degree (2013) from the Department of Chemical Engineering at China University of Petroleum (Huadong). The same year he started his PhD at the department of Chemical Engineering at Delft University of Technology. His research now is focused on the application of zeolite-based and MOF-mediated catalysts in heterogenous hydrogenation reactions.



Freek Kapteijn

Zhejiang Normal University. His research interest focuses on the interplay of catalysis and engineering, comprising structured and multifunctional catalysts, adsorption, separation and (catalytic) membranes. He has co-authored over 600 publications in peer-reviewed journals and book chapters.

As Billy Ocean once sang “*when the going gets tough, the tough get going*” and so it does apply to MOFs in catalysis. MOFs may not only be envisioned as catalytic solids themselves, but also as precursors for developing highly stable materials with special catalytic properties through the so-called MOF mediated synthesis (MOFMS). This synthetic technique makes use of the MOF as a sacrificial template. Under a certain atmosphere and at high temperature the framework collapses, the organic linker is completely or partially burnt away or carbonized (converted into a carbon matrix)¹⁵ and the metal ions/clusters become mobile within the newly created matrix or just evaporate. The essence of MOFMS lies not only in its simplicity but also in its versatility. Its variants embrace a whole range of approaches giving place to different types of materials. Depending upon the heat treatment and, in some cases further post-treatment(s), MOF precursors may lead to the formation of bulk and supported catalytic materials bearing unprecedented properties (Fig. 1).

In recent years, the synthesis of nanostructured metal oxides (MOs) has been a matter of intense investigation, with special emphasis on controlling the shape, composition and porosity of the resulting MOs.²⁷ The MOFMS has come to light as a promising technique for tailoring metal MOs in a simple manner. Calcination in air is the usual approach for the preparation of metal oxides from MOFs, but acid leaching and other heat treatments are also applied. Carbon materials may act as catalysts and catalyst supports. The main hurdle in the case of carbons as both support and catalyst lies in the development of novel methods able to tailor the porosity and morphology of carbonaceous nanostructures.²⁸ MOFMS yields well-structured hierarchical carbons by removing the metal phase during or after the carbonization of the MOF by volatilization at high temperature or acid leaching, respectively.



Jorge Gascon

Jorge Gascon (1977) received his MSc in Chemistry in 2002 and his PhD in Chemical Engineering in 2006, both at the University of Zaragoza (Spain). Since 2013 he has been ‘Anthonie van Leeuwenhoek Professor’ of Catalysis Engineering at TU Delft. His research interests include fundamental aspects and applications of new nano-structured materials and composites. He has co-authored over 170 publications, several patents and book chapters. He has been the recipient of several fellowships and awards, including VENI (2010), VIDI (2013) and ERC Starting (2013) personal grants and the 2013 ExxonMobil Chemical European Science and Engineering Award.



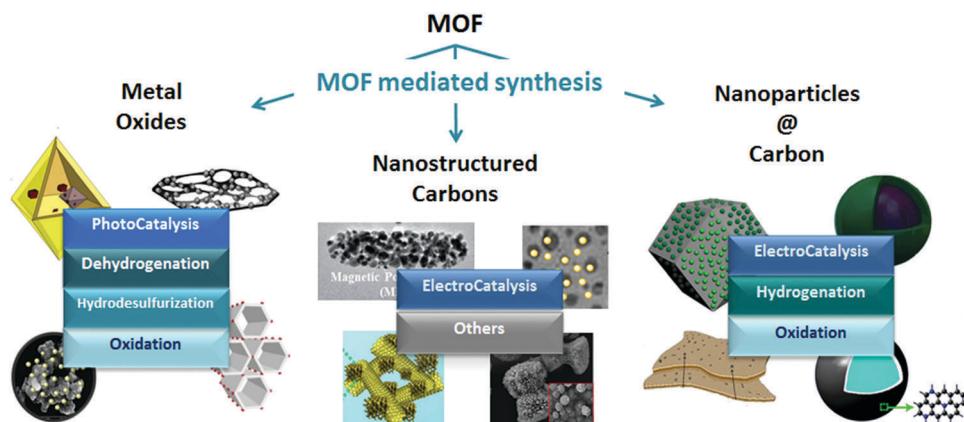


Fig. 1 Examples of nanostructures obtained through MOF mediated synthesis and their application in catalysis.^{16–26}

When it comes to supported nanoparticles, the tendency of nanoparticles to grow into larger crystallites is an impediment for stable catalytic performance.²⁹ Carbonizing the MOF in an inert atmosphere leads to a highly dispersed metal phase encapsulated in a porous carbon matrix, the spatial restriction created by encapsulation minimizing the sintering of the metal nanoparticles. As in the case of nanostructured carbons, the conventional strategy to obtain nanoparticles@carbon (NP@C) consists of carbonizing the MOF in an inert atmosphere at high temperature. However, contrary to the approach for carbons, in this case the metal phase remains retained in the resulting material. Therefore, during the pyrolysis, a part of the organic ligand decomposes and evaporates to the gas phase, whereas the other part becomes carbonized in the framework.

As recently highlighted by Sun *et al.*³⁰ and Xia *et al.*,³¹ MOFMS has been extensively applied during the last few years in the field of electrochemistry in both electrochemical energy storage and conversion. MOFMS has been thoroughly applied in supercapacitors, batteries and fuel cells. Moreover, over the past couple of years, the MOFMS technique has also been applied in the field of heterogeneous catalysis, the main topic of this review article.

Herein we highlight the main properties of MOFMS as a novel technique for the preparation of heterogeneous catalysts. Before reviewing the different examples from the literature, we first bring to the attention of the reader important considerations to be taken into account in the design of new solids *via* MOFMS. After the scene has been set, we thoroughly review the different examples from the literature. We do so by classifying these materials according to the metal in the framework of the pristine MOF. Last but not least, we share our personal assessment of the most promising catalytic materials and catalytic applications and the most important issues that still need to be addressed. This review aims at giving a critical overview of the current situation for the MOFMS in heterogeneous catalysis thus providing guidelines for the design of new generations of catalytic materials using MOFs as sacrificial agents.

2. General considerations about MOF mediated synthesis

Although the organic linker plays an essential role in the outcome of the final catalyst, it is the nature of the metal that governs the MOF mediated synthesis and determines the catalytic process the MOF mediated catalyst will be applied in. In short, the parent MOF is selected according to the metal(s) in its framework, whereas the organic linker and the synthesis conditions are envisioned as the tools that allow optimizing the final properties of the MOF mediated catalyst. Accordingly, once the MOF decomposes and envisaging the MOFMS as an essential heat treatment, the following parameters are key in determining the final properties of the resulting MOF derived catalyst:

(a) The Gibbs free energy (ΔG) of a reaction, such as oxidation of metals, quantifies the thermodynamic driving force necessary to make a reaction “work” under given conditions. A negative value for ΔG indicates that oxidation can proceed spontaneously without external inputs, while a positive value indicates that it will not. Widely used in extractive metallurgy, the Ellingham diagram (Fig. 2) plots the formation free energy of oxides *versus* temperature.³² The lower the position of a metal in the Ellingham diagram, the greater is the stability of its oxide and besides, the intersection of two lines in this diagram implies an oxidation–reduction equilibrium. Therefore, reduction using a given reductant is possible at temperatures above the intersection point where the ΔG line of that reductant is lower on the diagram than that of the metallic oxide to be reduced. In MOFMS, the carbon formed during the heat treatment acts as the reducing agent and reduction of metal oxides is carried out by a carbothermic reaction. Likewise, when the number of electrons involved is the same, the Gibbs free energy of oxidation follows exactly the opposite trend to the reduction potential of the corresponding metal ion for different metals. Accordingly, Das *et al.* quantified for the first time that metal ions with a standard reduction potential of -0.27 V or higher, such as Co, Ni and Cu, present in MOFs always form pure metal nanoparticles



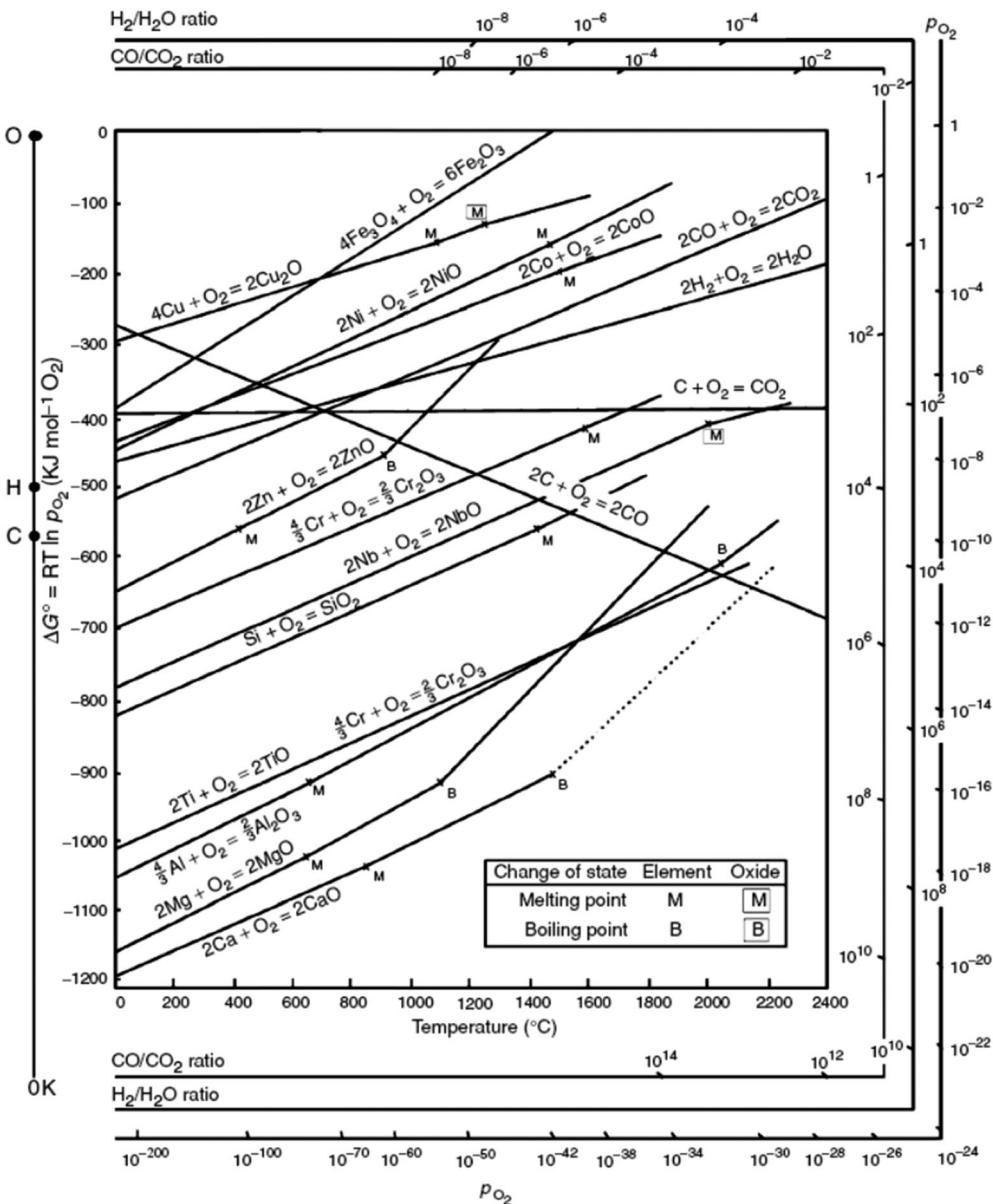


Fig. 2 Ellingham-diagram for the formation of oxides based on their standard free energy of formation over temperature.³⁴

during thermolysis in N₂ up to 900 °C, whereas metal ions with a reduction potential lower than -0.27 V, such as Mg, Al, Mn, Zn or Cr tend to combine with the oxygen present in the organic part of the MOF and form metal oxide nanoparticles during the same thermolysis treatment.³³

(b) Another property of utmost importance for MOFMS is the Tamman temperature of the actual metal species, which is approximately half the melting point of the metal species itself (Table 1). The exact definition of the Tamman temperature is that at which the atoms or molecules of the solid acquire sufficient energy for their bulk diffusion. Otherwise, the Huttig temperature which is approximately one third of the melting point of the metal species refers to surface species having sufficient mobility to begin to agglomerate and sinter. In MOFMS, the Tamman temperature of the metal species is

typically above the temperature required to make the organic framework collapse. Therefore, once the framework is decomposed it is according to their Tamman temperature that atoms start colliding and forming the corresponding nanoparticles, which, in line with thermodynamics, consist of metals, metal oxides, metal carbides, metal oxocarbides and/or a mixture of them. Increasing the temperature boosts the buildup of larger particles at the expense of smaller ones which is better known as the sintering process. In short, the Tamman temperature of different metal species is responsible for delimiting the temperature range in which the MOFMS must proceed if control over nanoparticle size is to be achieved. Likewise, the vapour pressure of the formed metallic species or mixture of them needs also to be taken into consideration in order to get rid of or retain the metal species in the final catalytic material.

Table 1 Melting and boiling points of different metals³⁵

Metal	Melting point, °C	Boiling point, °C
Al	660	2519
Cr	1907	2671
Co	1495	2927
Cu	1085	2560
Fe	1538	2861
Mg	650	1090
Mn	1246	2061
Ni	1455	2913
Ru	2334	4150
Ti	1668	3287
V	1910	3407
Zn	420	907
Zr	1855	4409

3. MOF mediated synthesis of heterogeneous catalysts

(a) Aluminium (Al)-MOFs

Aluminium is the third most abundant element in the Earth's crust (after oxygen and silicon) and the most abundant metal.³⁶ Among the different transition aluminas known, γ -alumina (γ -Al₂O₃) is perhaps the most important with direct applications as a catalyst and catalyst support in the automotive and petrochemical industries.³⁷ The usefulness of this oxide is due to a favourable combination of its textural properties, such as surface area, pore volume, and pore size distribution and its acid/base characteristics, which are mainly related to surface chemical composition, local microstructure, and phase composition.

The low position of Al in the Ellingham diagram indicates the high stability of its oxide even at high temperature and in the presence of carbon. The melting point of Al₂O₃ is 2072 °C, its boiling point is 2977 °C and the reduction potential of Al³⁺ is -1.66 V. Accordingly, bare pyrolysis of Al-MOFs should yield basically Al₂O₃ (reduction potential below -0.27 V³³) in a carbon matrix (high boiling point) at least up to 2977 °C, when it starts to evaporate. Therefore, if required, acid leaching applies for the removal of this metal oxide, leading to Al free carbons.

Due to their thermal and chemical stability, Al-containing MOFs, such as the well-known Al-MIL-53, Al-MIL-101-NH₂ and CAU-1 are subjects of intensive research.³⁸ Al-MOFs used in the synthesis of carbon supports for catalysis include CAU-1 and MOF-253. Prepared for the first time by Ahnfeldt *et al.*,³⁸ the 3D microporous framework Al-CAU-1 ($[\text{Al}_4(\text{OH})_2(\text{OCH}_3)_4(\text{H}_2\text{N}-\text{bdc})_3] \cdot x\text{H}_2\text{O}$) (bdc = 1,4-benzenedicarboxylate) is formed with two types of cages, distorted octahedral and distorted tetrahedral, with effective accessible diameters of approximately 1 and 0.45 nm, respectively. CAU-1 exhibits high porosity and thermal stability up to 360 °C. In this manner, reaction of AlCl₃·6H₂O with 2,2'-bipyridine-5,5'-dicarboxylic acid (H₂bpydc) affords Al(OH)(bpydc) (MOF-253) which displays a BET surface area of 2160 m² g⁻¹ being stable up to 400 °C.³⁹ The Al-MOF mediated synthesis reported so far includes the preparation of carbon supports for electrocatalysis and coupling reactions.

In electrocatalysis, Afsahi *et al.* confirmed the potential of Al-MOFs as templates to prepare efficient fuel cell electrocatalysts.⁴⁰ Firstly, they synthesized desolvated Al-MOF-253 and subsequently they introduced PtCl₂(CH₃CN)₂ together with acetonitrile. The mixture was heated at 85 °C for 72 h and the resulting solid was collected by filtration and immersed in acetonitrile for 3 days. Finally, the product was collected by filtration and dried at 150 °C for 12 h under vacuum. To prepare PEMFC electrocatalysts, the obtained Pt-MOF materials were heat treated at temperatures ranging between 700 and 1050 °C at 1 °C min⁻¹ for 4 h under Ar. The membrane-electrode assemblies (MEAs) made from the obtained Pt nanoparticles (7–10 nm) in carbon were tested as both an anode and a cathode in a H₂/air single cell fuel cell. As the anode, the most promising electrocatalyst (carbonized at 950 °C) demonstrated an open circuit voltage of 970 mV and a power density of 0.58 W mg_{Pt}⁻¹ comparable to the commercial electrode power density (0.64 W mg_{Pt}⁻¹) at 0.6 V in a single cell test. At the cathode, a power density of 0.38 W mg_{Pt}⁻¹ at 0.6 V was achieved. Although the results show that this rather simple technique allows the production of electrocatalysts with performances comparable to the commercial materials, the authors propose further improvements by (i) changing the heating inert atmosphere to a reactive gas such as ethylene or acetylene, possibly allowing better preservation of the original atomic dispersion of Pt in the MOF and (ii) starting with MOFs containing other metals such as Fe or Co in order to replace the highly expensive Pt and Pt groups metals (PGMs).

Quantum chemistry calculations predict an electrocatalytic behaviour of Co₉S₈ similar to that observed for a Pt electrode. In view of this, Zhu *et al.* prepared highly active Co₉S₈/CNS catalysts for ORR from aluminum-based MIL-101-NH₂ and Co/S precursors.⁴¹ MIL-101-NH₂ was selected as a prototypical host for the encapsulation of thiourea (TU) and cobalt chloride to form the catalyst precursor, due to its facile, economical and scalable synthesis. The as-obtained Co(II)TU@MIL-101-NH₂ sample was finally heated in Ar at 600 °C (or 700, 800, 900, 1000 °C) for 5 h with a heating ramp of 10 °C min⁻¹. The residual Al component and other impurities were removed by immersing the samples in a HF (20 wt%) solution for 24 h. The authors found that the resulting Co₉S₈@CNS material carbonized at 900 °C exhibited remarkably comparable ORR catalytic activity, and superior long-term stability and methanol tolerance compared with commercial 20 wt% Pt/C catalyst under alkaline conditions. They associated the outstanding electrocatalytic performance of the catalyst for the ORR with the unique honeycomb-like open structure with high surface area and pore volume, proper degree of graphitization, and high content of the active species combined with their synergistic interactions.

The palladium-catalyzed Suzuki–Miyaura coupling of aryl halides with aryl boronic acids is one of the most powerful methods for constructing biaryl structures, which are important units in pharmaceuticals, herbicides, and natural products.⁴² On the other side, N-decorated carbon materials enhance the mechanical and energy-storage properties, and show high stability as supports for noble metal nanoparticles.⁴³ In view of this, Zhang *et al.* successfully fabricated a novel catalyst based



on Pd nanoparticles supported on N-doped nanoporous carbon from Al-based MOFs with high catalytic activity for the Suzuki-Miyaura coupling reactions at room temperature.⁴⁶ The as-prepared CAU-1 was carbonized at 800 °C for 10 h in N₂ giving N-doped nanoporous carbon (NPC) and subsequently Pd nanoparticles were immobilized on the N-doped NPC by impregnation. Additionally, for the preparation of aluminium-free N-doped NPC-Pd, prior to Pd impregnation, N-doped NPC was treated with 10% HCl for 5 h, giving Al free N-doped NPC-Pd. Catalytic results indicated that N-doped NPC-Pd could catalyze the reaction efficiently even at room temperature within 1 h to afford the corresponding biphenyl in 97% yield with EtOH/H₂O as the solvent, compared to 78% yield obtained with the commercially available 5% Pd/C catalyst under the same conditions. The reaction did not take place in the absence of a catalyst nor using N-doped NPC as a catalyst, which indicated that aluminium oxide has no catalytic activity for the coupling reaction. However, for the aluminium-free N-doped NPC-Pd only 69% yield of biphenyl was obtained which demonstrated that the presence of aluminium oxide in N-doped NPC might be helpful to improve the dispersion of Pd nanoparticles.

In the aforementioned examples, although for different catalytic purposes, the authors successfully prepared carbon materials for supporting noble and non-noble metal particles from Al-MOFs. Although the presence of aluminium and aluminium oxide are thought to improve the dispersion of Pd nanoparticles for coupling reactions, no further explanation on the mechanism is given. Besides, no clear effect of Al is reported in the electrocatalytic performance of MOF derived catalysts. In this sense, there is a gap in Al-MOF derived catalysis. Filling this gap would imply (i) deeper analysis on the role of Al₂O₃ in improving the metal dispersion of noble metal nanoparticles and (ii) further research on carbon free nanostructured Al₂O₃ synthesis from MOFs through calcination in air.⁴⁴ However, for the latter and based on the wide availability of mesoporous aluminas the next question must be posed: is it really worth applying MOF mediated synthesis to this end?

(b) Chromium (Cr)-MOFMS

Chromium is found mainly in nature as chromite. The melting points of Cr and Cr₂O₃ are 1907 °C and 2435 °C, respectively, and their boiling points are 2671 °C and 4000 °C. Besides, the reduction potentials of Cr³⁺ to Cr²⁺ and of Cr³⁺ to Cr are -0.40 V and -0.74 V, respectively. Although Cr oxidation states range from +6 to +2, its most stable states found in the environment are +6 and +3. Whereas Cr(III) is believed to be an essential element, hexavalent chromium (Cr(VI)) is acutely toxic, a proven mutagen and carcinogenic heavy metal pollutant,⁴⁵ and is believed to be the second most common inorganic contaminant after lead.⁴⁶

Probably the best-known application of chromium in catalysis applies to ethylene polymerization or the Phillips polymerization process. The Cr(VI)/SiO₂ Phillips catalyst, patented in 1958 by Hogan and Banks, is a highly versatile system accounting for the production of about half of the world's market⁴⁷ of high

density (HDPE) and also linear low density (LLDPE) polyethylene, making it one of the world's most important industrial catalysts for over half a century.⁴⁸ In addition, Cr based catalysts are used in high temperature water gas shift,⁴⁹ dehydrogenation of propane and isobutane⁵⁰ and organic transformations.⁵¹

Regarding Cr-MOFs and firstly reported in 2005 by Férey *et al.*,⁵¹ MIL-101(Cr) is a chromium terephthalate MOF with the molecular formula Cr₃F(H₂O)₂O[O₂C-C₆H₄(CO₂)₃-nH₂O (where *n* is ~25) and stable up to 275 °C in air. The large pores (2.9 and 3.4 nm) and high BET area (>3000 m² g⁻¹) with a huge cell volume together with the coordinatively unsaturated open metal sites that can be subjected to diverse post-synthesis functionalization or guest encapsulation, and excellent hydrothermal/chemical stability, make MIL-101 particularly attractive for different applications including selective gas adsorption/separation, energy storage and heterogeneous catalysis for oxidation, hydrogenation, condensation, degradation and coupling reactions.⁵²

Consequently, Cr-MOFMS has been applied in the catalytic dehydrogenation of isobutane. In this industrially important route for producing isobutene, a vital component for the synthesis of octane number boosters for unleaded gasoline, the Cr₂O₃/Al₂O₃ catalyst has been applied on a commercial scale for years.⁵³ A few years ago, Zhao *et al.* studied the reactivity of isobutane dehydrogenation over a series of non-ordered mesoporous chromia/alumina catalysts promoted with K₂O using the MIL-101(Cr) metal-organic framework as a molecular host and chromium precursor.⁵⁰ They impregnated MIL-101 with aluminium isopropoxide (Al(i-OC₃H₇)₃) and KOH as the aluminium and potassium sources, respectively, and calcined the resulting material in air at different temperatures ranging from 600 to 900 °C for 4 h obtaining K₂O-Cr₂O₃/Al₂O₃ catalysts with large specific surface areas and high pore volumes. They found that the pore texture, structural phase, reducibility and surface concentrations of Cr³⁺ and Cr⁶⁺ species over the catalysts depended on the chromia loadings and calcination temperature. They proposed that the Cr³⁺ species were mainly the active sites and the catalytic selectivity depended on the surface Cr³⁺/Cr⁶⁺ value over the catalyst. The K₂O addition to the catalyst slightly decreased the specific surface area and the surface Cr³⁺/Cr⁶⁺ value but it greatly improved the isobutene selectivity and reduced the deactivation rate. Accordingly, the catalyst with 1.5 wt% K₂O and 10 wt% Cr₂O₃ loadings calcined in air at 800 °C for 4 h was found to exhibit the highest isobutane conversion activity with high isobutene selectivity and a more stable dehydrogenation activity than the conventional supported chromia catalyst prepared by the conventional impregnation method.

Another noteworthy example of MOF derived Cr catalysts is given by Qiu *et al.* who worked on the controlled growth of dense and ordered MOF derived nanoparticles on graphene oxide (GO) for organic transformations.⁵¹ In their synthesis procedure, GO was mixed with metal ions for a long time before the organic ligand was added enabling a sufficient coordination between the surface groups of GO with metal ions, providing dense and homogeneous nucleation nodes for the formation of MOFs.



Thus, MIL-101-GO hybrids with high-density and ordered MOF particles were achieved, which featured different sizes and morphologies as compared to the parent MOFs. Cr-containing/rGO nanomaterials were obtained after pyrolysis of a MIL-101-GO hybrid at 900 °C under Ar for 8 h. The higher activity of this material in ethylbenzene oxidation compared to the merely carbonized MIL-101 and MIL-101-GO prepared by other reported method was attributed to the smaller size of Cr_3C_2 particles dispersed by rGO, leading to more exposed active sites for organic transformations. The authors postulate that this method might bring new opportunities for the synthesis of growth-controllable MOF-GO and metal-containing/rGO nano-hybrids for advanced functional applications by combining both fascinating materials.

All in all, both the high number of Cr catalysed processes and the versatility of MIL-101(Cr) regarding synthesis, design and applications⁵² offer a large amount of possibilities for MOFMS. Regarding the preparation of CrO_x , it is worth re-emphasizing the potential hazard that handling chromium(vi) oxide (CrO_3) constitutes during the synthesis, loading, and unloading of the catalyst. Keeping this in mind and proceeding accordingly, the stability of MIL-101 up to 275 °C in air establishes the lower limit of temperature for its calcination in air, whereas the upper one needs to be established according to the targeted porosity of the final metal oxide depending on its catalytic application. Otherwise, for the preparation of supported nanoparticles and back to Cr's properties, the reduction potentials of Cr^{3+} to both Cr^{2+} and metallic Cr predict that the heat treatment in an inert atmosphere of Cr(III)-MIL-101 leads mainly to Cr(II), in the form of Cr_3C_2 at temperatures close to 1200 °C according to the Ellingham diagram.

(c) Cobalt (Co)-MOFMS

In nature cobalt is found in cobaltite, skutterudite and erythrite whereas most cobalt is obtained as a by-product of nickel refining. The high position of Co in the Ellingham diagram evidences the low stability of its oxide. This means that the pyrolysis of Co-MOFs will lead to metallic Co by carbothermal reduction at low temperature close to 200 °C. The melting point of metallic Co is 1495 °C, its boiling point is 2927 °C and the reduction potential of Co^{2+} is -0.27 V, thus confirming the theory by Das *et al.*³³

The use of Co in different catalytic applications mainly depends on the following facts: (i) the oxidation-reduction properties of cobalt and its ability to demonstrate several valencies I + II + III with easy electron transfer between these states, (ii) cobalt's ability to form complexes by accepting atoms from other molecules, (iii) cobalt chemicals in solution and in polymerisation systems can decompose to give more than one ion to take part in catalysis, and (iv) solid cobalt compounds have vacancies in their crystal lattices which can take part in catalysis. Therefore, cobalt is used as a catalyst in oxidation reactions such as the conversion of xylene to terephthalic acid; the steam reforming of ethanol for hydrogen production proceeds over cobalt oxide-base catalysts;⁵⁴ cobalt-based catalysts are used in reactions involving carbon monoxide being the main catalyst

in the low-temperature Fischer-Tropsch process for the hydrogenation of carbon monoxide into liquid fuels;⁵⁵ hydroformylation of alkenes often uses cobalt octacarbonyl as a catalyst,⁵⁶ besides, the hydrodesulfurization of petroleum uses a catalyst derived from cobalt and molybdenum, and furthermore, cobalt is also used as a catalyst in electro- and photocatalytic processes.

ZIF-67 is the Co-MOF preferred choice when it comes to Co-MOFMS. ZIF-67 ($\text{Co}(\text{HmIM})_2$) is isostructural to ZIF-8, and is formed by bridging 2-methylimidazolate anions and cobalt cations resulting in a sodalite (SOD) topology with a pore size of about 0.34 nm.⁵⁷ ZIF-67 nanocrystals can be synthesized in aqueous solutions at room temperature, thus not only prohibiting the usage of toxic organic solvent, but also decreasing the cost of preparation of MOF materials. Furthermore, the concentration of reagents is proved to play an important role in controlling particle size, and diluting the synthesis solution can increase the particle size of ZIF-67.⁵⁸

In photocatalysis, work done so far on Co-MOFMS includes dye-sensitized solar cells (DSCs), which have the ability to convert solar energy into electricity. DSCs are generally composed of a dye-sensitized photoanode, an electrolyte, and a counter electrode (CE), the latter being crucial to the photovoltaic performance of DSCs.¹ Typically, a Pt CE enables DSCs to achieve high power conversion efficiency (PCE), but the high cost of Pt severely limits the overall development of the cells. Against this background, Jing *et al.* developed a facile one-step approach using ZIF-67 as a sacrificial template in the synthesis of a CE catalyst for DSCs.⁵⁹ ZIF-67 was pyrolyzed in N_2 following a heating rate of 5 °C min⁻¹ and maintained at 850 °C for 2 h, yielding ZIF-67-850 which was found to be the optimum calcination temperature, producing the highest degree of graphitization with ameliorated surface wettability. ZIF-67-850 exhibited good electrochemical performance and the PCE attained was very close to that observed for the Pt CE in the liquid I_3^-/I^- redox couple electrolyte. Characterizing the structure of the products indicated that cobalt nanoparticles were embedded in an N-doped graphitic carbon matrix while cobalt and cobalt oxide nanoparticles were exposed on the external surface of the carbon. The authors attributed the excellent performance of ZIF-67-850 to the synergistic effects between the Co and CoO coupled with the nitrogen doped graphitic carbon. They finally concluded that the electrocatalytic performance, cycle stability and low-cost of the preparation make this novel catalyst a promising candidate to replace Pt-based materials in solar cell applications.

Pt nanoparticles supported on Vulcan™ XC-72R microporous carbon are the most common catalysts used to accelerate the sluggish oxygen reduction reaction (ORR) at the cathode of a proton exchange membrane fuel cell (PEMFC). There are several examples of Co-MOFMS in electrocatalysis, with special emphasis on the ORR. As a matter of fact, Ma *et al.* identified a Co-imidazolate framework $\text{Co}(\text{Im})_2$ (Im = imidazolate) described elsewhere⁶⁰ as an efficient precursor for this reaction.⁶¹ The sample started to demonstrate ORR activity after heating it at 600 °C and its optimal performance was achieved when the sample was pyrolyzed at 750 °C with an onset potential comparable to



the best cobalt-based non-PGM catalysts.⁶² The combined analysis of TEM, XPS and XAFS led the authors to believe that upon thermal activation, the catalytic sites were constructed by Co–N interactions through pyridinic or pyrrolic N species retained in the carbon.⁶³ This model explains why the ORR activity decreases with higher pyrolysis temperatures, as more N is lost with increasing carbonization, and comprises the starting point for prospective work on the topic.

Xia *et al.* obtained similar results by pyrolyzing Co-based ZIF-67 in an Ar atmosphere with temperatures in the range of 600–900 °C.⁶⁴ The current density of the ORR was also maximized in this case for the sample carbonized at 750 °C, explained by the optimum graphitic N atoms witnessed by XPS.⁶³ Moreover, the authors studied the effect of the crystallite size of the MOF precursor and showed that as the size decreased, the onset potential and kinetic current density increased. Finally, they concluded that the high activity of the nano-electrocatalyst is attributed to the fully exposed nano-structure together with good mass- and electron-transport properties.

In other work, Xia *et al.* proved that MOFs can be coated on highly ordered mesoporous carbon prior to pyrolysis to provide a final catalyst structure with improved mass and electron transfer characteristics.⁶⁵ This procedure comprised the coating of a Co benzimidazole MOF on a CMK-3 mesoporous carbon and subsequently pyrolyzing it at 700 °C, and optionally oxidizing the Co phase in air at 150 °C. The heat treatment resulted in finely dispersed Co agglomerates over the 3D pore structure with sizes around 15–30 nm, without causing pore blockage due to the interconnected carbon matrix (Fig. 3). The different heat treatment procedures and additional acid leaching of Co generates a tunable set of ORR catalysts that gives highly relevant insight into the Co core and carbon shell, next to very efficient ORR catalysts in terms of activity and stability.

With their work Wang *et al.* provided further insight into Co-MOFMS for electrocatalysis by analysing three representative MOF structures (ZIF-67, ZIF-8 and $\text{Co}_2(\text{bdc})_2(\text{dabco})$ (dabco = 1,4-diazabicyclo[2.2.2]-octane)) as the catalyst precursors for the ORR.⁶⁶ Such choices allowed them to study how the metal/ligand combination affected the ORR performance. The ORR catalysts

were prepared *via* pyrolysis in Ar of the MOFs at different temperatures ranging from 600–1000 °C for 10 min and with 10 °C min⁻¹ heating ramp. They found that among the three MOF precursors, the ZIF-67 derived catalyst exhibited the best ORR activity in both alkaline and acidic electrolytes. They suggested that Co atoms with stronger interactions with the carbon matrix, for instance, those directly bonding with the pyrrolic/pyridinic N, were retained in the nanosized voids and these N-coordinated Co sites inside the nanoscale pores served as the active ORR sites. Furthermore, the best ORR performance was obtained when the porosity and the graphitic structure of ZIF-67 were optimized at 900 °C pyrolysis temperature, and the Co content was reduced from 30 wt% Co to 4.7 wt% by acid leaching in HCl due to the removal of Co particles in the nanopores which were blocking the CoN_x active sites. Notably, the acid leached sample exhibited even more positive onset and half-wave potentials than those of Pt/C in alkaline electrolytes, as well as better stability and tolerance to methanol crossover. The overall performance of the acid leached sample is among the best of the contemporary noble metal free ORR catalysts in both alkaline and acidic electrolytes which the authors ascribed to the highly dispersed active CoN_x sites in the nanoporous conductive system.

Kong *et al.* worked on the one-step pyrolysis of a unique “cage-in-cage” Co-MOF to prepare nitrogen-doped carbon cubes with numerous embedded metallic Co nanoparticles and tested them for the ORR.⁶⁷ In their case, CPM-24 Co-MOF was selected since (i) its highly symmetric isotropic cubic symmetry was expected to contribute to the uniformity in the distribution of heteroatoms in a carbon matrix, (ii) it consists of a unique nested $\text{Co}_{24}@\text{Co}_{48}$ nearly spherical cage-within-cage configuration which best meets the desired structural features that could facilitate the creation of a core–shell configuration and (iii) it also consists of the Co–N linkage which allows the entrapment of nitrogen sites for an enhanced activity. CPM-24 Co-MOF was synthesized *via* a modified indium assisted multi-component self-assembling method previously reported by the same authors.⁶⁸ CPM-24 Co-MOF cubic crystals were heated for 2 h under Ar, with a heating rate of 3 °C min⁻¹ and the final temperature reached between 550 and 900 °C depending

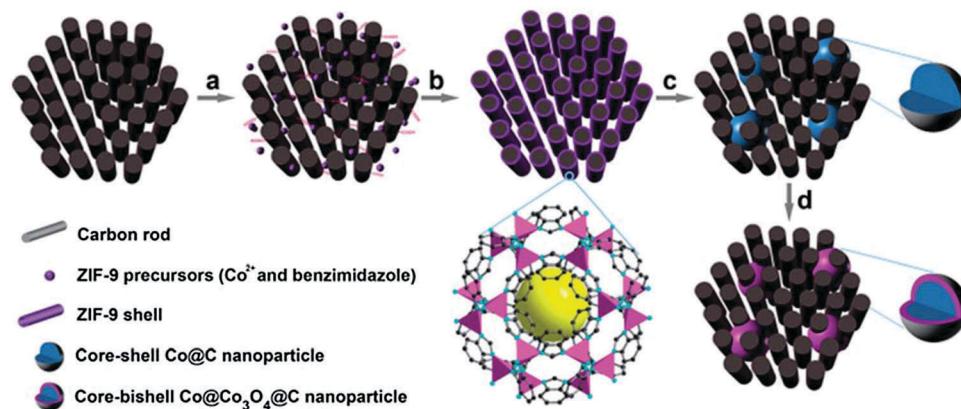


Fig. 3 Schematic illustration of the synthesis of $\text{Co}@\text{Co}_3\text{O}_4@\text{C}-\text{CM}$: (a) functionalization of the CM with carboxylate groups, (b) growth of the MOF on the CM, (c) heat treatment under Ar flow and (d) oxidation in air at 90 °C. Reproduced from ref. 65 with permission from The Royal Society of Chemistry.



on the sample. As a result, ferromagnetic Co nanoparticle-embedded and nitrogen-doped carbon composites were obtained. The sample prepared at 700 °C demonstrated excellent methanol-tolerance and showed high ORR activity comparable to the commercial Pt/C (10 wt%) catalyst in alkaline medium and also showed higher durability than Pt/C. The authors proved through long-time etching by HF that Co nanoparticles in the composites were essential in enhancing the ORR electrocatalytic performance which in principle contradicts what Wang *et al.*⁶⁶ previously suggested when attributing the catalytic activity mainly to Co-N_x moieties. In contrast, Kong *et al.* believe that nitrogen-doped carbon activated by Co nanoparticles played the important role in ORR performance of their MOF derived Co electrocatalysts.

To rationally design the reaction interface for an oxygen evolution reaction (OER), an optimal electrode structure involving high-performance catalysts is essential in reducing the overpotential, promoting the reaction kinetics, and enhancing the specific activity for the OER, thus improving the energy efficiency.⁶⁹ To this end, a lamellar Co naphthalenedicarboxylate MOF was used by Ma *et al.* to grow nanowire arrays (NAs) on a Cu substrate plate, followed by carbonization at 600 °C.⁷⁰ Specifically, the material was first heated from room temperature to 400 °C with a ramp rate of 1 °C min⁻¹, and the temperature was kept at 400 °C for 2 h, which was further increased to 600 °C and stabilized for another 4 h. Pyrolysis of the layered crystalline MOF structure created slit-like pores and the array of thick nanowires provided a large accessible surface area. Noticeably, the OER current of Co₃O₄-C-NA largely exceeded that of a similarly loaded IrO₂/C coated on Cu despite the slightly lower onset potential of IrO₂/C, featuring a much better catalytic performance of Co₃O₄-C-NA and also in terms of durability. Moreover, the carbonized MOF on Cu proved to be a better electrode configuration than calcined MOF on Cu, evidenced by electrochemical impedance spectra and polarization curves.

Leaving electrocatalysis behind, there are a large number of examples that endorse the application of MOF derived cobalt catalysts in conventional heterogeneous catalysis in oxidation and hydrogenation reactions.

The low-temperature oxidation of CO has drawn tremendous interest due to its wide applications such as cleaning air, lowering automotive emission, and detecting trace amounts of CO. Bao *et al.* targeted the preparation of Co oxides from MOFs and demonstrated a novel “*in situ* self-reduction” route for the fabrication of a nanocomposite consisting of Ag/Co₃O₄ from Prussian blue analogues (PBAs) without using any reductant.⁷¹ After calcining the uniform Ag₃[Co(CN)₆] nanocubes at different temperatures in air, porous Ag/Co₃O₄ nanocomposites with abundant well-knit Ag/Co₃O₄ nanojunctions were obtained. The prepared samples exhibited excellent catalytic activity and stability for catalytic CO oxidation and the activity was maintained for more than 18 h, even after the catalyst was exposed to air for two months. Without any pretreatment, Ag/Co₃O₄ catalysts prepared by these authors showed outstanding catalytic performance compared with other oxide supported-Ag catalysts reported in recent years. The presence of abundant and highly dispersed Ag/Co₃O₄ junctions in the nanocomposites was

suggested to be responsible for the excellent catalytic performance. On the junctions, O₂ and CO adsorb on Ag and Co₃O₄, respectively, and react with each other more easily and besides, Ag particle aggregation could be avoided during a catalytic reaction. Coming back to ZIF-67 mediated synthesis for supported Co nanoparticle preparation, Wang *et al.* recently disclosed the preparation of an efficient metallic Co-based catalyst for low-temperature CO oxidation, where the Co was embedded in porous carbon.⁷² ZIF-67 was pyrolyzed at 600 °C for 10 h in Ar and the obtained Co/C-600 catalyst was used for CO oxidation under dry conditions yielding 100% CO conversion at 0 °C (T100). The apparent activation energy of CO oxidation over the Co/C-600 catalyst was comparable to those of the Co₃O₄,⁷³ Pd⁷⁴ and Au⁷⁵ catalysts. When comparing the activity of Co/C-600 to Co supported on activated carbon, it was found that the light-off temperature decreased by 78 °C for Co/C-600. The catalytic activity over the Co/C-600 catalyst remained unchanged after 24 h of time on stream at room temperature, thus indicating that Co/C-600 was highly stable. However, in the presence of small amounts of moisture, Co₃O₄ deactivates quickly even under ambient temperatures. Interestingly, catalytic stability tests for Co/C-600 under wet gas conditions showed improved moisture tolerance as compared to other Co-based materials and moreover, the activity of the catalyst could be fully restored upon a mild heating treatment in an oxidative atmosphere.

Air pollution by volatile organic compounds (VOCs) has been a serious environmental problem in past decades. Among the VOCs, benzene is an important chemical feedstock and a gasoline ingredient.⁷⁶ For complete oxidation of benzene to CO₂ and H₂O in this case, Li *et al.* studied tricobalt tetraoxide-supported palladium catalysts derived from MOFs.⁷⁷ Porous polyhedron (PP) Co₃O₄ catalysts were prepared through one-step calcination of ZIF-67 in a muffle oven at an elevated temperature (250, 350, or 550 °C) at a ramping rate of 1 °C min⁻¹ and held for 1 h. Finally, Pd/Co₃O₄-PP-X (X = 250, 350, or 550 °C) were prepared by a conventional impregnation method and subjected to calcination at 200 °C for 4 h. Among the MOF derived catalysts obtained at different temperatures, Co₃O₄-PP-350 exhibited the highest activity for the complete catalytic conversion of benzene. This catalyst also showed better performance than that of Pd supported on Co₃O₄ nanoparticles (NPs) prepared by a hydrothermal method and calcined at the same temperature. They resolved that (i) the proper porosity and nanoparticle size of the catalyst could be controlled by adjusting the calcination temperature, (ii) the surface adsorbed oxygen, which is associated with the PdO_x species, is crucial for improved catalytic performance and (iii) the reducibility of the catalyst has a significant effect on the catalytic activity for the oxidation of benzene. As a consequence, the porous structure could expose more PdO_x species on the surface of the catalyst, and the electron transfer from Pd to O for Pd/Co₃O₄-PP-350 might make the surface adsorbed oxygen more active, rendering Pd/Co₃O₄-PP-350 as the best catalyst for the complete oxidation of benzene.

Direct oxidation of alcohols to esters with molecular oxygen is an attractive and crucial process for the synthesis of fine chemicals.



The heterogeneous catalysts used so far are based on noble metals or require the addition of base additives. Therefore, the development of reusable non-noble metal based catalysts for the oxidative direct esterification of alcohols under mild conditions is an attractive and challenging subject in both green chemistry and organic synthesis. To this purpose, Zhong *et al.* developed a simple, cost effective, and environmentally friendly protocol for direct aerobic oxidative esterification of alcohols, using a novel nitrogen doped graphite enclosed cobalt material from ZIF-67 as a catalyst.⁷⁷ The conversion of ZIF-67 into Co@C-N was performed by direct thermal treatment under a flow of Ar at 800 °C for 8 h with a heating rate of 1 °C min⁻¹. The strong coordination interaction between Co and N atoms in the parent MOF allowed a stepwise collapse of the MOF structure during slow-heating procedures to prevent a serious aggregation of Co. The resulting carbon–nitrogen composite played an important role in isolating the Co nanoparticles, which were surrounded tightly by graphitized carbon. The catalytic system featured a broad substrate scope for aromatic and aliphatic alcohols as well as diols, giving their corresponding esters in good to excellent yields at room temperature and under atmospheric conditions without the assistance of any base additives. Moreover, the catalyst was easily recovered due to its good magnetic properties and the reactivity could be fully restored for up to five runs when the reused catalyst was treated in H₂ at 400 °C for 1 h. In a similar manner, Zhou *et al.* recently reported a similar route to generate Co NPs with surface-oxidized CoO species uniformly incorporated into N-doped porous carbon by one-step pyrolysis of ZIF-67, as a hard template.⁷⁸ The pyrolysis was carried out at different temperatures and times under N₂ and it was found that the MOF pyrolyzed at 700 °C for 3 h showed the best properties. Thus, the resultant Co–CoO@N-doped porous carbon nanocomposite as an inexpensive, stable and magnetically recyclable catalyst exhibited excellent catalytic performance for direct homo-and cross-coupling esterifications of primary alcohols under mild conditions, with 1 bar of O₂ as an environmentally friendly oxidant. The special structure of the nanocomposite and the synergistic effect between Co and CoO NPs were proposed by these authors to be responsible for the superior activity compared to other related catalysts. In addition, the N-doped porous carbon scaffold not only stabilized the NPs but also greatly facilitated the accessibility and adsorption of substrates to the active sites and the diffusion of products.

The development of a highly efficient and environmentally sound methodology for the synthesis of amides from inexpensive and abundant feedstocks is highly desirable in both academic research and industrial applications. In this regard, Bai *et al.* addressed a novel oxidative amidation of aldehydes using MOF mediated heterogeneous Co-based catalysts.⁷⁹ The as-prepared Co^{II}MOF (Co₉(btc)₆(tpt)₂(H₂O)₁₅, tpt = 2,4,6-tris(4-pyridyl)-1,3,5-triazine) was heated at 600 °C for 8 h with a heating rate of 1 °C min⁻¹ in Ar. With 4-methylbenzaldehyde as a model substrate and DMF as the solvent, Co@CN600 gave *N,N*-4-trimethylbenzamide in 90% yield at 80 °C. Moreover Co@CN600 could be easily separated by using an external magnetic field and could be reused at least five times with only

a slight loss of activity. Control experiments using (i) homogeneous Co(NO₃)₂, (ii) the parent Co^{II}MOF, (iii) CoO or Co₃O₄ nanoparticles, (iv) metallic Co nanoparticles (20–30 nm), (v) activated carbon and (vi) a physical mixture of Co and carbon as the catalysts highlighted the importance of synergic interactions between the C–N composite and Co nanoparticles in determining the activity of the Co@C-N materials in the oxidative amidation reaction. With the optimized reaction conditions in hand, the authors also proved that the MOF derived catalysts were highly active in oxidative amidation of a wide range of aldehydes with formamides, affording the corresponding amides in good to excellent yields under mild reaction conditions. The performance was mainly attributed to the resulting small Co nanoparticles embedded in the N-doped carbons highly dispersed with an average size of *ca.* 7 nm. Three distinct peaks were observed in the N1s spectra of Co@C-N600, suggesting the presence of three kinds of coordination environments for N atoms: pyridine type nitrogen bonded to a metal (advantageous to prevent a serious aggregation of Co during pyrolysis of the MOF), pyrrole-type nitrogen (usually found in the carbonized nitrogen-containing organic materials and which might confer the porous N-doped carbons with a remarkably enhanced chemical reactivity due to their extended electronic structures) and N typical for ammonium species.

The ability to transport electrons has been identified as the key property of the resultant catalyst in advanced oxidation reactions. Andrew *et al.* adopted MOFMS, and combined ZIF-67 with graphene oxide (GO), in order to enhance electron transport and use GO as a co-catalyst to activate peroxyomonosulfate (PMS).⁸⁰ The ZIF-67/GO nanocomposite was carbonized in N₂ at 600 °C for 6 hours, and subsequently washed with water and ethanol thoroughly to obtain the final product, magnetic cobalt-graphene (MCG). MCG was used as a heterogeneous catalyst to activate PMS for the decolorization of acid yellow in water. MCG exhibited a unique morphology and porous structure, and sufficient saturation magnetization also allowed it to be readily collected from water after the decolorization reaction. Moreover, this material exhibited an enhanced catalytic activity in the activation of PMS compared to the merely carbonized ZIF-67. To evaluate the long-term catalytic activity of MCG, a 50-cycle decolorization test was performed and the regeneration efficiency remained the same over 50 cycles, showing its stable and effective catalytic activity.

Hydrogen as a clean energy has attracted wide attention. Xing *et al.* focused their efforts on the preparation of multilayer core–shell composites from Co-MOFs loaded on reduced graphene oxide (rGO) sheets for hydrogen generation from NaBH₄.⁸¹ The as-obtained Co-MOF–poly vinylpyrrolidone (PVP)–GO black powder was further heated with a ramp of 5 °C min⁻¹ to 700 °C in N₂ for 1 h yielding Co@N-CG and further heat treated at 200 °C for 24 h in air yielding Co@CoO_x@N–carbon–rGO (Co@CoO_x@N-CG). Additionally, the authors prepared Co@CoO_x@CG, Co@CoO_x@N-C and Co@CoO_x@C, in the absence of PVP, GO and both of them, respectively. The Co@carbon and Co@carbon–rGO core–shell



structures were obtained with the formation of cores from metal ions and carbon shells from carbonization of ligand molecules. Controllable oxidation of Co cores to Co shells generated multilayer core–shell structures, namely $\text{Co}@\text{CoO}_x@\text{carbon}$. In hydrogen generation through hydrolysis of NaBH_4 , multilayer core–shells, $\text{Co}@\text{CoO}_x@\text{carbon}$ possessed higher catalytic activity with lower activation energy compared to single core–shells. The authors explained the improved catalytic activity by (i) the synergy effects between Co and CoO_x on the interface region and (ii) the carbon shells or N-doped carbon shells protected the Co or $\text{Co}@\text{CoO}_x$ cores from growing, aggregating, or breaking up for excellent stability. Moreover, the rGO sheets facilitated the magnetic momentum transfer and improved the flexibility of these core–shell catalysts. The authors deem these core–shell structures anchored on rGO sheets to be an excellent alternative for the design of effective heterogeneous catalysts involving energy industrial routes.

Likewise, transition metal-catalyzed transfer hydrogenation protocols are convenient and alternative green chemical methods to traditional energy-consuming hydrogenation processes, besides being able to provide high atom efficiency and generating advantageous economics.^{82,83} The use of hydrogen donor reagents such as alcohols in transfer hydrogenation reactions can avoid the use of autoclaves and high-pressure hydrogen, being highly relevant for industrial applications. Not long ago, Long *et al.* hinted at the transfer hydrogenation of unsaturated bonds in the absence of base additives catalyzed by MOF derived cobalt heterogeneous catalysts.⁸⁴ They selected $[\text{Co}(\text{bdc})(\text{ted})_{0.5}]$ MOF as a template due to the presence of triethylenediamine basic sites which are deemed to be favorable for transfer hydrogenation reactions. In the transfer hydrogenation of acetophenone to phenethanol as a model process using isopropanol as both reductant and solvent, the $\text{Co}@\text{C–N}$ materials prepared under longer thermolysis times and higher temperatures exhibited higher activity and selectivity, thus the final selected catalyst was obtained by pyrolysis in argon of the Co MOF at 900 °C for 15 h. Multiple advantages of the proposed catalytic system included a remarkable versatility not only for C=O hydrogenation but also for C=C , $\text{C}\equiv\text{N}$, and N=O bonds, an environmental-benign protocol and the use of a magnetically separable non-noble metal catalyst in a simple and safe reaction setup. Both mechanisms proposed in the literature for the transfer hydrogenation with base additives suggest that the presence of bases facilitates the formation of metal hydrides in the transition metal.⁸⁵ Therefore, the authors assumed that basic sites on $\text{Co}@\text{C–N}$ might play a similar role to that of base additives to facilitate the transfer of protons to Co nanoparticles to form Co metal hydride species, thus promoting the production of the desired hydrogenation products. Combination of high efficiency, versatility and recyclability as well as mild reaction conditions in the absence of bases and gaseous hydrogen makes this system an attractive alternative pathway for various hydrogenation processes.

Generally, different anilines are prepared by reduction of their corresponding nitro substrates. Nevertheless, selective hydrogenation of a nitro group in the presence of other reducible substituents is challenging. Wang *et al.* synthesized

chemoselective nitrogen-doped carbon supported Co MOF derived catalysts for the selective hydrogenation of nitroarenes starting from ZIF-67.⁸⁶ Both conversion and selectivity decreased with an increase in pyrolysis temperature of the MOF, suggesting that the presence of nitrogen, which decreases with increasing temperature, might play an important role in this catalytic system. Shen *et al.* focused on the hydrogenation of nitrobenzene and proposed a facile and efficient approach to fabricate non-noble metal@noble metal core–shell catalysts by using Co-MOF as a sacrificial template.¹⁸ $\text{Co}@\text{Pd}$ core–shell NPs embedded in a N-doped carbon matrix ($\text{Co}@\text{Pd}/\text{CN}$) were prepared using ZIF-67 and $\text{Pd}(\text{NO}_3)_2$ as the precursor and Pd source, respectively. First, ZIF-67 was pyrolyzed at 600 °C under Ar for 8 h and the obtained Co NPs were used as a sacrificial template to reduce an aqueous $\text{Pd}(\text{NO}_3)_2$ solution *via* galvanic replacement reaction resulting in the coverage of the MOF-derived Co NPs by an ultrathin Pd shell. The highly exposed Pd atoms on Co nanoparticles resulted in a highly active, extremely stable catalyst for nitrobenzene hydrogenation. The $\text{Co}@\text{Pd}/\text{NC}$ showed much higher hydrogenation activity than MOFs supporting noble NPs, thus opening a new avenue for MOF-templated non-noble@noble metal core–shell catalysts that could far surpass the traditional MOFs supporting noble NPs in catalytic properties. Moreover, as a novel approach, the one-step pyrolysis of ZIF-67 carried out by Ma *et al.* produced an N-doped porous carbon for the tandem dehydrogenation of ammonia borane and hydrogenation of nitro compounds at room temperature.⁸⁷

Most of the examples dealing with Co-MOFMS for heterogeneous catalyst preparation refer to metal nanoparticles in carbon where pyrolysis under an inert atmosphere is applied. Ample data are available on this topic when compared with other metal based MOFs. Besides, due to the ease of synthesis and low price, ZIF-67 has been the most studied MOF in MOF derived catalyst preparation. ZIF stands for a zeolitic imidazolate framework and refers to a sub-family of MOFs with imidazolate as the linker. The structures of ZIFs are similar to conventional aluminosilicate zeolites and thus, generally display properties that combine the advantages of both zeolites and MOFs, such as ultrahigh surface areas, unimodal micropores, high crystallinities, abundant functionalities and exceptional thermal and chemical stabilities.⁸⁸ In MOFMS, the main feature of ZIF-67 derived catalysts is the content of N which results in advantageous properties in different catalytic processes due to the resulting (i) nitrogen-doped carbon, (ii) $\text{Co}-\text{N}_x$ moieties and (iii) synergistic interactions between the C–N composite and Co nanoparticles. Thermodynamics predict that the carbothermal reduction of Co ions in the framework of the MOF leads mainly to metallic cobalt. Metallic cobalt catalyses the graphitization of surrounding carbon which in some processes comprises the main limitation for the use of these catalysts, since the hardly accessible nanoparticles might lead to diffusion limitations of the reactants to the active Co sites during the process.

All things considered and since in some processes the nature of the active sites still remains under debate, full understanding and control of the synthesis process calls for further research on Co-MOFMS, including ZIF-67-MOFMS.



(d) Copper (Cu)-MOFs

The applications of copper (Cu) and Cu-based nanoparticles, which are based on this earth-abundant and inexpensive metal, have generated a great deal of interest in recent years, especially in the field of catalysis.⁸⁹ Higher in the Ellingham diagram than Co, the reduction potential of Cu²⁺ is +0.34 V and the melting and boiling points of Cu are 1085 °C, and 2560 °C, respectively. The catalytic application of Cu is broad, including electrocatalysis, photocatalysis, and gas-phase thermal catalysis.

HKUST-1 (MOF-199) is one of the most cited and employed Cu-MOFs because of its large surface area, high pore volume, high chemical stability and easy synthesis. First reported in 1999 by Chui *et al.*,⁹⁰ HKUST-1 is made up of copper nodes with 1,3,5-benzenetricarboxylic acid (btc) struts between them. In the framework of HKUST-1, Cu(II) ions form dimers, where each copper atom is coordinated by four oxygens from btc linkers and water molecules,⁹¹ forming face-centered-cubic crystals that contain an intersecting three-dimensional (3D) system of large square-shaped pores (9 Å by 9 Å). Cu-bdc's such as MOF-2 are other examples of Cu-MOFs with markedly high surface area, which makes them superior candidates for gas separation and sieving applications.⁹² The coordination mode in MOF-2 is the same as the coordination mode in HKUST-1. However, in contrast to HKUST-1, the architecture of MOF-2 is based on stacking lamellar phase sheets of Cu₂(bdc)₂ in two dimensions.⁹³ MOFs with open coordination sites, like HKUST-1 and MOF-2, will likely be scrutinized for possessing both homogeneous and heterogeneous catalytic behaviour.⁹²

Both metal oxides and nanoparticles@carbon have been reported using mainly HKUST-1 as the precursor of electro- and photocatalysts as well as for CO oxidation and the transformation of alcohols and aldehydes.

In photocatalysis, Li *et al.* described a facile synthetic method for the preparation of Cu/TiO₂ photocatalysts with hollow structures by treating HKUST-1 at 140 °C in the presence of ascorbic acid (AA) and ethylene glycol (EG) as solvent and TiO₂ hollow particles.²⁰ The released Cu²⁺ cations were simultaneously reduced by AA to produce Cu nanoparticles when they penetrated through the TiO₂ shells. The AA-reduced Cu/TiO₂ hybrid structures exhibited enhanced charge separation and hydrogen production performance. Likewise, Mondal *et al.* constructed MOF-derived Cu/CuO@TiO₂ photocatalysts where the copper species were adsorbed on anatase.¹⁹ Titanium isopropoxide (Ti(ipro)₄) was added to the Cu-bdc MOF in an ethanol/water mixture and then the amorphous material was calcined at 350 °C for 5 h in a N₂ atmosphere. The obtained mesoporous Cu/CuO@TiO₂ showed a considerably higher rate of H₂ production than conventional CuO loaded TiO₂, attributed to the presence of surface deposited Cu⁰ species and the small size of the heterojunction between CuO and TiO₂, which facilitated interfacial charge carrier transfer from TiO₂ nanoparticles.

In the field of electrocatalysis, palladium is a potential candidate to substitute platinum as a catalyst for the ORR since on the one hand, Pd is a member of the Pt group metals and thus its catalytic ability is close to that of Pt and on the other hand, the Pd abundance in the earth crust is higher than that of Pt.

Further, addition of transition metals such as Cu⁹⁴ to the catalytic system enhances the catalytic performance and decreases precious-metal loading. In this line, Luo *et al.* successfully synthesized a PdCu alloy on carbon (PdCu/C) using HKUST-1 as the catalyst precursor.⁹⁵ The synthesis procedure consisted of adding K₂PdCl₆, sodium acetate, HKUST-1 (Basolite C300, Sigma-Aldrich) and carbon black (heat treated Vulcan XC-72) into methanol and purging by nitrogen in an ice-water bath (ca. 2–4 °C) for 20 min under continuous stirring. Subsequently, the system was sealed under CO in an ice-water bath for 2 h and then sealed under N₂ at room temperature for 12 h under stirring. Finally, the product was filtered and rinsed with ultra-pure water and the obtained black powder was dried under air at 60 °C overnight. The improved ORR activity obtained on the MOF derived PdCu/C compared to PdCu/C synthesized from a metal chloride precursor (CuCl₂) was correlated with Pd surface modification induced by alloying, further proved by CO-stripping voltammograms. Compared with the commercial Pt/C Johnson Matthey (JM) catalyst, the ORR mass activity of PdCu/C was lower in acid medium, whereas it was around 3-fold higher in alkaline solution. In view of these results, the authors claim that using HKUST-1 precursors for the synthesis of PdCu nanoalloyed NPs, not only extends the application of MOFs in the field of electrocatalysis, but also brings a new methodology in advanced non-Pt active centers for the ORR. Besides, work done by Raoof *et al.* highlights the good behaviour of HKUST-1 derived Cu/nanoporous carbon composites for the hydrogen evolution reaction (HER).⁹⁶ The Cu/NPC composite catalyst was synthesized by direct carbonization of HKUST-1, the latter prepared according to a previously reported procedure.⁹⁷ HKUST-1 was then heat treated in N₂ at 550 °C, followed by 6 h at 900 °C yielding the Cu/NPC composite. Electrochemical measurements of the Cu/NPC on a glassy carbon electrode indicate improvement of HER catalysis which they attribute to the existence of Cu, CuO, and Cu₂O species together with improved surface area, and the expected hydrogen adsorption properties of the Cu/NPC composite.

Copper based catalysts look also promising for substituting the limited available noble metals in the preferential oxidation of CO (PROX) in the presence of hydrogen, which is one of the most promising ways to reduce the CO content of effluents from reforming units.⁹⁸ Not yet envisioned as a synthesis method itself, Zamalo *et al.* realized that HKUST-1 in CO oxidation was not active by itself, but after 'in situ' activation in the reaction atmosphere at temperatures above 230 °C, highly dispersed and highly active CuO nanoparticles were produced.⁹⁹ The transformation involved a gradual segregation of CuO nanoparticles from the MOF, simultaneously with a loss of MOF crystallinity. Moreover, the incorporation of a Ce precursor in the network of HKUST-1 with subsequent activation yielded a highly dispersed mixture of CuO and CeO₂ nanoparticles with a high degree of interaction that was highly active in CO oxidation above 230 °C. Later, as a proof of concept, Liu *et al.* used the same MOF and TiO₂ whiskers¹⁰⁰ as supports to prepare a well-dispersed and size-controllable CuO/TiO₂ catalyst derived from the thermolysis of the Cu-btc/TiO₂ composite



through calcination in air.¹⁰¹ The CuO/TiO₂ catalyst behaved excellently in CO oxidation and the strategy could be translated to other supports, with CuO/Al₂O₃ showing excellent activity.

With focus on the transformation of alcohols, Niu *et al.* synthesized a hybrid catalyst for the catalytic reduction of 4-nitrophenol consisting of non-noble Cu/Cu₂O NPs supported on porous carbon using HKUST-1 as a sacrificial template.¹⁰² After thoroughly mixing HKUST-1 with a phenol formaldehyde resin as an additional carbon source, the resulting solid product was carbonized under a N₂ atmosphere at 600 °C for 4 h. The Cu/Cu₂O/C composite demonstrated excellent catalytic activity for the reduction of 4-nitrophenol in the presence of NaBH₄, comparable with noble metal based catalysts, such as SiO₂/Fe₃O₄-C/Au,¹⁰³ Au-Ag/GO,¹⁰⁴ and Au/CuO.¹⁰⁵ Moreover, the synthesized catalyst could be reused for at least five cycles due to its good stability. The authors claimed that Cu/Cu₂O NPs were responsible for the rapid reduction of 4-nitrophenol while the carbon matrix could prevent aggregation of Cu/Cu₂O NPs providing high surface-to-volume ratio and chemical stability. In another recent work, a new catalytic system based on HKUST-1, TEMPO and NMI (TEMPO: 2,2,6,6-tetramethylpiperidine-N-oxyl; NMI: *N*-methylimidazole) was developed by Kim *et al.* for aerobic alcohol oxidation.¹⁰⁶ As-prepared HKUST-1 was calcined under Ar at 800 °C for 6 h with a ramping rate of 5 °C min⁻¹. For the reaction, a test tube was loaded with equimolar amounts of the catalyst and TEMPO and evacuated and backfilled with oxygen. The alcohol and NMI with acetonitrile were then added to the mixture and kept under stirring at 70 °C under an O₂ balloon. Simple pyrolysis of HKUST-1 afforded a robust and chemically inert copper–carbon nanocomposite (Cu@C) to form the Cu@C/TEMPO/NMI system which exhibited outstanding performance for the aerobic oxidation of alcohols to their corresponding aldehydes. Furthermore, this catalyst system revealed a broad scope for the substrate and excellent reusability for the aerobic oxidation of alcohols compared to the non-carbonized HKUST-1/TEMPO/NMI system.

In the catalytic hydrotreatment of furfural, Cu catalysts have strong ability to polarize the C–O bond while leaving the furan ring unaffected and furthermore, the appropriate addition of nickel might enhance their performance due to the synergistic effect of bimetal.¹⁰⁷ Accordingly, Wang *et al.* addressed the preparation of CuNi bimetallic nanoparticles embedded in a carbon matrix (CuNi@C) from MOFs and applied them in the selective conversion of biomass-derived furfural to cyclopentanone.¹⁰⁸ For the preparation of CuNi@C bimetallic catalysts with the Ni/Cu molar ratio from 0 to 2.5, a predetermined amount of Ni(NO₃)₂·6H₂O ethanol solution was added to as-prepared HKUST-1. The mixture was kept under stirring for 24 h and then dried at 80 °C overnight. The resulting solid was calcined in N₂ for 2 h at 500 °C to obtain the CuNi@C catalyst. The authors found that CuNi_{0.5}@C showed the best catalytic performance when compared to other Cu/Ni ratios and also when compared to CuNi_{0.5} prepared by the classical precipitation method. In addition, CuNi_{0.5}@C could be reused four times with good activity and stability. The enhanced structure of the composite, the nano-scaled metallic copper and nickel

without oxide impurity highly dispersed in a porous carbon matrix are thought to be responsible for the outstanding catalytic performance in furfural conversion to cyclopentanone.

In summary, despite the strong background on the applications of bulk Cu in various fields the use of Cu NPs is restricted by their inherent instability under atmospheric conditions, *i.e.* their sensitivity for oxidation. The possible modification of the chemical and physical properties of these nanoparticles using different synthetic strategies and conditions and/or *via* post-synthetic chemical treatments has been largely responsible for the rapid growth of interest in these nanomaterials and their applications in catalysis.⁸⁹ In this sense MOFMS reinforces the idea of creating a chemically inert metal or metal oxide–carbon composite by pyrolysis of MOFs as a powerful general strategy for preparing stable and reusable copper heterogeneous catalysts. Nevertheless, the high reduction potential for Cu²⁺ means that when calcined in an inert atmosphere the vast majority of copper appears in its metallic form. Metallic copper strongly promotes the graphitization of the surrounding carbon which leads to poor accessibility of reactants to active Cu sites during the catalytic process.

(e) Iron (Fe)-MOFMS

Iron minerals are ubiquitous in nature and there is abundant availability of them at minimal cost.¹⁰⁹ Haematite is the most common iron containing ore, but iron is widely distributed in other minerals such as magnetite and taconite. The melting point of metallic Fe is 1538 °C, its boiling point is 2861 °C and the reduction potential of Fe³⁺ and Fe²⁺ are -0.04 and -0.41 V, respectively. The magnetic nature of iron enables its easy separation from aqueous media; its high boiling point ensures high iron loading after pyrolysis and the formation of carbides due to C incorporation in the Fe lattice and prompts the metallic, carbidic, oxidic dynamic system in the pyrolysis. As in the case of Co and Cu, Fe also catalyzes graphite formation in the carbons seen as graphite layers.

Iron catalysis embraces Fe/N/C systems for ORR catalysis for fuel cell applications, FeC for FTS, Fe-mixed-C systems for nitro/chloro conversion and separation from waste water (dye and oil experiments), among others.

Recently, de Krafft *et al.* reported a simple MOF-templated strategy for the synthesis of crystalline octahedral nanoshells combining the desired photocatalytic properties from both types of oxides (Fig. 4).¹¹⁰ In their work, they coated MIL-101(Fe) particles with an amorphous shell of titania by acid-catalyzed hydrolysis and condensation of titanium(IV) bis(ammonium lactato)-dihydroxide (TALH) in water,¹¹¹ followed by calcination in air at 550 °C for 16 h. They found the resulting crystalline Fe₂O₃@TiO₂ nanoparticles to have interesting photophysical properties as they exhibited photocatalytic hydrogen production from water using visible light, while neither component alone was able to do so. Kim and Yoon¹¹² reported the pyrolysis of MIL-100(Fe) and the effects of furfuryl alcohol (FA) added to the MOF-pores acting as a carbon precursor. They displayed the ferromagnetic properties of the polycrystalline Fe₂O₃@C system, allowing for facile particle separation from a slurry phase. Elemental analysis



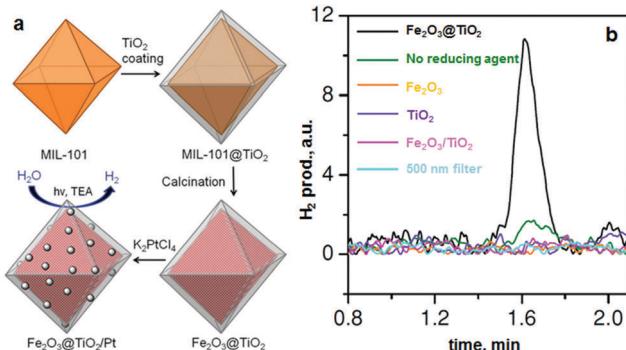


Fig. 4 (a) MOF-templated synthesis of $\text{Fe}_2\text{O}_3@\text{TiO}_2$ by coating MIL-101(Fe) with TiO_2 followed by calcination, and its use for photocatalytic hydrogen production after depositing Pt particles; (b) the H_2 peaks in GC analysis of the headspace using 0.5 mg of different catalyst samples. From de Krafft *et al.* 2012.¹¹⁰ Copyright © 2012 by John Wiley Sons, Inc. Reprinted by permission of John Wiley & Sons, Inc.

revealed very minor hydrogen content, clearly indicating the conversion of the organic linkers toward carbonaceous materials upon pyrolysis. Indeed, the highest carbon loading of the composite material was obtained when fully soaking the MOF in FA, filling the interstitial sites of the MOF crystals in addition to the pores. Banerjee *et al.*¹¹³ pyrolyzed MIL-53(Fe) at 500 and 600 °C to produce Fe_3O_4 -carbon nanocomposites as recyclable adsorbent materials for removal and recovery of environmental pollutants. The sample pyrolyzed at 500 °C was highly hydrophobic due to the presence of small micropores around 0.8 nm in diameter and the absence of hydrophilic surface groups, much contrary to the sample pyrolyzed at 600 °C comprising larger mesopores with an average of 2 nm. This allowed for effective separation of oil/water mixtures using the super hydrophobic sample, achieving separation factor values over 40. The recyclability of the material is excellent, due to its magnetic character, retaining over 75% of the separation capacity after 9 cycles. Quin *et al.* reported the synthesis of CoFe_2O_4 nanocomposites from Fe-containing MOF (MIL-100-Fe) for phenol oxidation in aqueous solution using peroxymonosulfate as an oxidant.¹¹⁴ They immobilized cobalt nitrate hexahydrate into the pores of MIL-100 by incipient wetness impregnation and calcined it at different temperatures (400–600 °C). The resulting magnetic nanoparticles reduced phenol concentration by more than 95% after 120 min and the regenerated catalysts exhibited good performance and stability in phenol degradation. Likewise, Dong *et al.* reported the preparation of magnetic porous carbons (MPCs) as supports for Au and Pd noble metal NPs.²¹ They obtained the MPC through calcination of Fe-MIL-88A up to 700 °C under N_2 for 1 h. Accordingly, both Au/MPC and Pd/MPC nanocatalysts exhibited excellent catalytic activity for the reduction of 4-nitrophenol (4-NP) in comparison with most other reported catalysts.¹¹⁵ Moreover, the Pd/MPC catalyst showed high efficiency toward hydrodechlorination of chlorophenols. In both the synthesized catalysts, the MPC provided a large surface area and mesopores on which the Au and Pd NPs were finely dispersed. The catalysts could be easily recovered from the reaction mixture by using a magnet because of the

magnetic NPs encapsulated in the porous carbon, and could be conveniently reused. These methods are presented by the authors as an eco-friendly strategy to convert nitro- and chlorocompounds. A short while ago, Li *et al.* implemented MOFMS in the hydrogenation of nitro compounds by developing a highly efficient, low-cost, and magnetically recyclable $\gamma\text{-Fe}_2\text{O}_3@\text{porous carbon}$ catalyst.¹¹⁶ In their study, Fe-MIL-88A samples were thermally treated at different temperatures and times in N_2 to afford Fe-based NPs encapsulated inside porous carbon. The reduction of nitrobenzene to aniline in the presence of hydrazine hydrate ($\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$) was first used to explore the catalytic activity of different catalysts. As a result, the catalyst pyrolyzed at 500 °C for 1 h gave the best activity and could be readily recycled with an external magnet being reused at least 10 times without any loss of activity. They concluded that at this temperature the small $\gamma\text{-Fe}_2\text{O}_3$ NPs were well dispersed inside the porous carbon, which effectively limited the aggregation and growth of the high-density NPs and still facilitated the transportation of substrates, intermediates and products. Encouraged by the excellent performance of Fe-500-1 h in the reduction of nitrobenzene, a wide variety of functionalized nitro compounds were also investigated. As a result, most of the corresponding substituted anilines were obtained with high conversion and selectivity. Notably, halogen-substituted nitroarenes were effectively reduced to the corresponding haloaromatic amines without obvious dehalogenation. This catalyst also showed absolute selectivity in the reduction of nitroarenes in the presence of other functional groups, including amino, phenolic hydroxyl and alcoholic hydroxyl. Last but not least, it was also able to offer a high chemoselectivity in the reduction of substituted nitroarenes with quite challenging reducible functional groups, including nitrile, aldehyde and ketone, giving the only product of the corresponding anilines while the reducible functional groups were unchanged. Lee *et al.*¹¹⁷ pyrolyzed Fe-MIL-88-A and Fe-MIL-88-B in a range of temperatures between 600–1000 °C under nitrogen flow, creating magnetic nanoparticle-embedded porous carbon materials. The XRD results showed that the Fe phase shifted from a mix of Fe_2O_3 and Fe_3C at 600 °C to fully metallic Fe at 1000 °C. The porosity of the carbon materials went through an optimum at 800 °C, indicating pore collapse and decreasing surface areas at elevated temperatures. The two Fe-MOFs pyrolyzed at 800 °C were used for water treatment using methylene blue as a model compound, obtaining adsorption capacity values of 60.2 and 83.9 mg g⁻¹ for pyrolyzed MIL-88-A and MIL-88-B, respectively.

Additionally, MOFMS has been applied in various attempts to improve the catalysis of the oxygen reduction reaction for fuel cell applications. In order to obtain the desired Fe/N/C composites, different synthesis procedures have been followed. For example, pyrolysis of low-boiling temperature Zn-based MOFs produce highly porous carbons that provide excellent support surface for Fe nanoparticles. Similarly, Fe-based MOFs, often with amino-functionalized linkers, are pyrolyzed to produce Fe/N/C in a one-step synthesis. The main difference between the two synthetic pathways is the location of the Fe species. Lower surface loadings can be achieved through impregnation of the pyrolyzed MOF, however; only Fe-MOF pyrolysis results in the



formation of encapsulated Fe nanoparticles in the carbon matrix. Morozan *et al.*¹¹⁸ used the most complex MOFMS strategy comprising a first pyrolysis of FA impregnated ZIF-8 at 1100 °C, subsequent Fe and N functionalization by impregnation and ball milling, followed by secondary Ar pyrolysis at 1050 °C and even a third NH₃ pyrolysis at 950 °C. The various approaches generated metallic and carbide iron next to large fractions of FeN₄ species in a micro- and mesoporous carbonaceous matrix, of which larger mesopores did not further enhance the catalytic ORR activity. The boost in PEMFC performance after ammonia pyrolysis was ascribed to the increase in the surface area, interconnecting existing active sites for catalysis instead of increasing the active site density. Zhang *et al.*¹¹⁹ concluded that Fe species indeed are crucial to the catalytic performance, ruling out that the initial activity loss of the Fe/N/C catalysts is due to the Fenton reaction of Fe with H₂O₂. Furthermore, the kinetics of Fe/N/C and Fe-free N/C catalysts during the ORR were similar, revealing that micropore flooding through electro-oxidation of the carbon surface was indeed the main reason for the activity drop, in agreement with their previous study.¹²⁰

Mao *et al.*¹²¹ mixed MIL-100(Fe) with aminotetrazole and pyrolyzed the materials at temperatures between 600–900 °C to produce Fe–N–C composites. The products were washed with HCl to produce N-doped carbon shells encapsulating the metallic Fe nanoparticles, removing Fe₃C and Fe₂O₃ particles on the carbon surface. The treatment resulted in doubling of the micropore area, liberating cavities to trap oxygen while exposing active sites. The electrocatalytic properties of the materials showed that increasing pyrolysis temperatures resulted in increased current density and a positive shift of the onset potential during linear sweep voltammetry. The trend can be related to the facilitated electron transfer by the enhanced graphitic crystallinity witnessed in XRD. The highest catalytic activity can be achieved with the sample pyrolyzed at 800 °C (NC-800); the authors argue that higher pyrolysis temperatures decrease the N-doping and thus the number of active sites, in agreement with lowered N-content from EDS results. Cyclic voltammetry showed that NC-800 outperformed metal-free N-doped carbons, confirming the involvement of Fe in catalysis. Furthermore, the MOF-based catalyst showed improved stability during continuous ORR compared to the commercial Pt–C catalyst and comprised a higher methanol tolerance. In another study, Afsahi *et al.*¹²² used a sodalite-like Fe MOF consisting of benzene tris tetrazole linkers to obtain hybrid Fe–N–C after pyrolysis in the range of 700–1000 °C. The products were subsequently acid leached in 1 M H₂SO₄ and heat treated at the pyrolysis temperature under a NH₃ atmosphere. XRD of the pyrolyzed samples shows multi-crystalline Fe phases including Fe carbides, oxides, sulphides and nitrides, with all patterns showing much resemblance. Acid leaching removed the unstable Fe oxides and sulphides, and after NH₃ heat treatment Fe–N–C catalysts were produced. The particle size of the Fe spheres varies between 10–60 nm, which are dispersed in a highly porous cube-shaped carbon material. After acid washing, the Fe content drastically reduced from 40 wt% to around 5–10%, an increase in pyrolysis temperature

was found to further carbonize the structure, removing N species. The methodological approach allowed the authors to conclude that the ORR catalytic activity originates from the pyridinic and quaternary nitrogen species. Additionally, the performance of the catalysts in a fuel cell did not exceed the one from Dodelet's group,¹²³ and it is believed that the higher Fe content plays a prominent role in the reduced activity. Li *et al.*¹²⁴ pyrolyzed ball-milled MIL-101(Fe) mixed with melamine in a range of temperatures between 500–1000 °C to obtain Fe/Fe₃C embedded in graphite layers and carbon nanotubes. High temperatures above 700 °C and a melamine carbon precursor were necessary to produce the graphitic nature and to form the carbon nanotubes. Raman spectra confirmed the more graphitic nature of high temperature samples, nearly doubling the G-band over I-band intensity values. The ORR activity in terms of onset potential was very close to commercial Pt–C, and the electron transfer number from the Koutecky–Levich plots was estimated to be 3.6 compared to 3.7 for Pt–C. Zhao *et al.*¹²⁵ pyrolyzed nanocrystalline spindles of MIL-88B-NH₂ in Ar at 900 °C to produce very active ORR catalysts. The XPS results indicated that the amino-type nitrogen was completely converted into pyridine- and quaternary-type nitrogen species, forming a well-integrated Fe/N/C catalyst. The micro- and mesoporous matrix of the catalyst was found to consist predominantly of amorphous carbon, with layered graphite species surrounding the Fe nanoparticles. The power density of an alkaline direct methanol fuel cell increased 1.7 times when using the pyrolyzed MOF as compared to the commercial Pt/C catalyst. Furthermore, the MOF-derived catalyst showed superior stability toward methanol poisoning, whereas the current of the commercial Pt/C decreased by 80% upon introduction of 2 M methanol during chronoamperometric response measurements at 0.84 V in oxygen saturated 0.1 M KOH.

Recently, MOFs have been used as a precursor to produce high-potential Fe catalysts for the Fischer–Tropsch synthesis reaction (Fig. 5).¹²⁶ The Fischer–Tropsch synthesis (FTS) is a process for flexible production of key chemicals from synthesis gas; the main reaction being the hydrogenation of CO. Santos *et al.* employed the commercially available Fe–BTC as a precursor in MOFMS. Heat treatment of the MOF at 500 °C in an inert atmosphere resulted in the formation of a Fe@C composite that showed exceptional activity and stability in FTS. The key factor in catalytic performance consisted of maintaining the dispersion of the active phase. The pyrolysis of the MOF at low heating rates gently carbonized the framework while capturing the Fe in a carbonaceous matrix formed by the decomposition of the linker. With a pyrolysis temperature that is a factor of three lower than the melting point of the metal, sintering could be effectively reduced to synthesize highly dispersed Fe@C systems. Furthermore, the addition of a carbon source allowed for further tuning the porosity and Fe loading of the materials. Wezendonk *et al.* investigated the conversion of the Fe–BTC towards Fe@C by *in situ* XAFS and Mossbauer that revealed many intermediate Fe phases.¹²⁷ Starting from the Fe(III) oxo-clusters, the pyrolysis formed mixtures of Fe carbides and a Wüstite phase. After reduction



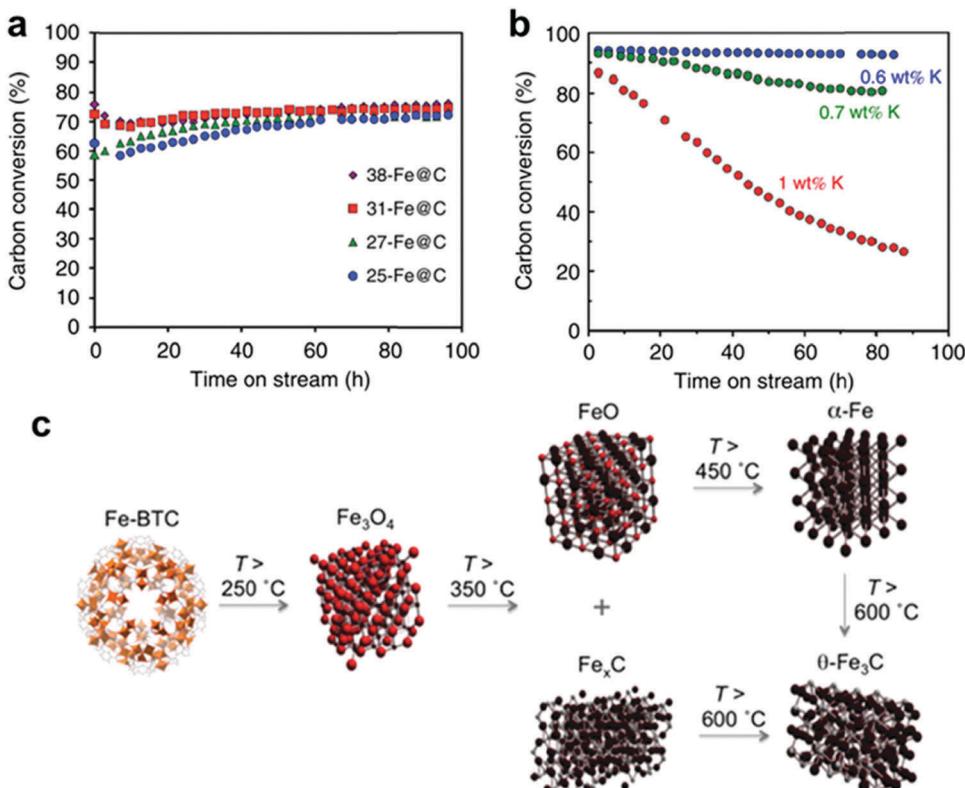


Fig. 5 (a) Time-on-stream evolution of CO conversion for unpromoted Fe@C catalysts with different Fe loadings; (b) time-on-stream evolution of CO conversion for K-promoted 38-Fe@C catalysts; and (c) Fe phase transformation during pyrolysis of Fe–BTC towards Fe@C. Reprinted by permission from Macmillan Publishers Ltd: *Nature Communications*,^{126,127} copyright 2015 Reprinted with permission from Wezendonk *et al.* 2016.^{126,127} Copyright 2016, American Chemical Society.

of the catalyst, the Wüstite was fully converted to metallic Fe and thus, upon syngas exposure creates highly active Hägg carbides. The system showed no deactivation, most probably due to the unique spatial confinement, witnessed with XPS and TEM. Hence, the catalyst combines a high loading of 40 wt% Fe with a high dispersion of which over 95% can be transformed into the active phase, resulting in the highest reported values for Fe-based FTS activity. Hence, the spatial confinement created by the porous carbon matrix on one hand allows for near-full transformation into an active phase and on the other hand prevents the metal NPs from sintering even under high-temperature FTS. Bing An *et al.*¹²⁸ used a MOF with dicarboxylic linkers instead of tricarboxylic MOFs to obtain larger external surface Fe particles to properly assess the active phase with microscopy. They found that the pyrolysis additionally leaves behind carbonates on the Fe surface, which is liberated from carbonate coverage during FTS. They confirm that the Hägg carbide phase is the active phase in HTFTS, and show microscope results of spent catalysts evidencing carbide surface growth on the large Fe oxide particles. Because of the slower active phase evolution, steady state conversion is achieved only over 60 h instead of 3 h for the system with much smaller particles.²² High activities were achieved at 300 °C with the activity correlating well with the thickness of the carbide shell on the large particles. Interestingly, a control experiment

employing pyrolyzed molecular Fe-oxo clusters showed little activity, underlining the importance of a framework-type precursor for pyrolysis and generating dispersed Fe particles. Wezendonk *et al.*¹²⁹ further investigated the effect of the MOF precursor using different MOFs while employing fixed pyrolysis conditions. It was found that under the same temperature, the different MOFs undergo a similar degree of carbonization of the framework by decarboxylation. The smallest Fe nanoparticles were obtained when pyrolyzing the most porous MOFs, confirming that the spatial distribution of the MOF is the key to produce catalysts with high dispersion and loading. Indeed, the performance could be related to the surface area of Fe, confirming the dispersion-activity relationship,²² while optimal hydrocarbon selectivity could be obtained with promoters such as K, Cu, Mn and S.

(f) Magnesium (Mg)-MOFMS

Magnesium is the eighth most abundant element in the Earth's crust and it occurs naturally only in combination with other elements, where it invariably has a +2 oxidation state.³⁶ The melting point of Mg is 650 °C, its boiling point is 1090 °C and the reduction potential of Mg²⁺ is –2.37 V. MgO is popular for its broad application as an industrial base catalyst and a catalyst support employed in aldol additions, Claisen–Schmidt condensations, transesterifications, and other basic catalysed reactions.¹³⁰

To the best of our knowledge, there is only one study reported to date dealing with Mg-MOF derived catalytic materials. In this example, a three dimensional framework with parallel hexagonal channels constructed from Mg^{2+} ions and 3,5-pyridine dicarboxylic acid was employed.¹³¹ In their work, Salehifar *et al.* developed a low-temperature synthesis procedure to obtain MgO nanorods *via* thermal decomposition of this Mg-MOF and evaluated its photocatalytic activity.¹³² The as-prepared Mg-MOF was treated in an autoclave under hydrothermal conditions at 200 °C for 6 h. The obtained product was washed with distilled water and absolute methanol several times and dried at 70 °C for 5 h to yield MOF derived MgO nanorods. In order to evaluate the photocatalytic activity of the MgO nanorods, the decolorization of methylene blue (MB) was studied. The authors found that in the presence of MOF derived MgO, the MgO nanorods obtained from MOFs have remarkable photocatalytic activity compared to other studies in the literature.¹³³ Here, MOF mediated synthesis brings not only simplicity but also decreases the costs and the temperature required in comparison with other similar techniques for synthesizing MgO.^{133–135}

To sum up, Mg-MOFs are still practically unexplored as catalyst precursors. However, the fact that one of the most fruitful methods for preparing mesoporous MgO was the “template” method, whose synthesis procedures require expensive templates and multiple steps besides being time-consuming¹³⁰ makes MOFMS an auspicious alternative technique for the synthesis of MgO. As in the case of Zn, since Mg^{2+} has one of the lowest reduction potentials, the preparation of nanostructured MgO from Mg-MOFs can be achieved by both direct pyrolysis in an inert atmosphere (MgO@C) and calcination in air (MgO) of the pristine MOF. Similar to that of Zn, the boiling point of Mg is considerably low, therefore one might think of using Mg-MOFs as templates for the preparation of hierarchical carbons by direct pyrolysis at around 1000 °C. However, looking at the Ellingham diagram, MgO with a boiling point of 3600 °C is only reduced by carbon at around 1850 °C thus making this preparation route of carbons far from energy efficient and thus non-cost effective.

(g) Manganese (Mn)-MOFMS

The chemistry of manganese is quite complex. Compounds containing manganese with formal oxidation states from -3 to +7 are known, however, under normal conditions, the +2 state is the most stable.¹³⁶ The melting point of Mn is 1246 °C, its boiling point is 2061 °C and the reduction potential of Mn^{2+} is -1.18 V. Therefore, in this case as well, pyrolysis in an inert atmosphere of the pristine Mn-MOF will lead to a mixture of Mn oxides in carbon whereas the calcination in air will generate the corresponding carbon-free metal oxide.

Due to their unique redox properties, low cost, and environment compatibility, manganese oxides are well known candidates for numerous catalytic applications¹³⁷ including decomposition of hydrogen peroxide, partial oxidation with hydrogen peroxide, oxidation of CO, methane, aromatic compounds, aqueous phenol in aqueous solution, NO–CO reaction, and chemical-looping combustion processes.¹³⁸

The only MOF derived Mn based catalyst reported so far in the literature appears in the work carried out by Peng *et al.*¹³⁹ These authors developed a novel methodology for the formation of large-pore mesoporous mixed valence Mn_3O_4 *via* the conversion of a Mn-MOF fabricated in ionic liquid (IL)–water mixtures for the degradation of MB. After IL and water removal, the Mn-BDC underwent thermolysis under N_2 with a heating rate of 5 °C min⁻¹ from room temperature to 400 °C. After reaching the target temperature, the material was calcined for 2 h and then naturally allowed to cool down to room temperature. In the presence of Mn_3O_4 , more than 99.7% MB degraded after only 1.5 h at 75 °C, compared to the reported behaviour for Mn_3O_4 crystals,¹⁴⁰ which needed 3 h to achieve a MB degradation >99.7% at 80 °C under similar concentration conditions.

The catalytic properties of manganese oxides depend on their structures, oxidation states and surface areas. To improve the catalytic properties of manganese oxides, therefore, these properties should be controlled. Manganese oxide structures are generally controlled through the synthesis conditions and they are usually supported on materials with high surface areas.¹³⁸ MOF mediated synthesis of Mn-MOFs to obtain mesoporous Mn oxides looks the most encouraging and straightforward approach for its application in catalysis, mainly in oxidation reactions and photocatalytic degradation of dyes. Here, the oxidation state of Mn in the pristine MOF, the atmosphere employed during the decomposition of the MOF together with the temperature of the heat treatment will play an essential role for controlling the size and the oxidation species present in the final catalyst due to the broad spectrum of Mn oxides available.

(h) Nickel (Ni)-MOFMS

Nickel is cheap, sufficiently active for certain reactions, and allows suitable catalysts to be economically produced.¹⁴¹ The melting point of Ni is 1455 °C, its boiling point is 2913 °C and the reduction potential of Ni^{2+} is -0.25 V. Nickel's position is just above palladium in the periodic table, and as a group 10 metal, it can readily catalyse many elementary reactions for which palladium or platinum are active as well. Because of these commonalities, nickel is often viewed solely as a low-cost replacement catalyst for cross-coupling reactions. However, this common misconception is clearly refuted by the numerous and diverse nickel catalysed reactions reported in the literature,¹⁴² such as, oxidative dehydrogenation of ethane,¹⁴³ partial oxidation¹⁴⁴ and both dry¹⁴⁵ and steam reforming of methane,¹⁴⁶ methanation of CO¹⁴⁷ and hydrotreating processes¹⁴⁸ among others. Besides RANEY® nickel is used in a large number of industrial processes mainly focused on hydrogenation.^{149,150} Furthermore, particularly remarkable is the recent development of nickel based homogeneous or organo-nickel chemistry.¹⁵¹

p-Aminophenol is an important intermediate in the preparation of several analgesic and antipyretic drugs. It is also used as a corrosion inhibitor in paints and as an anticorrosion-lubricating agent in fuels for two-stroke engines and in the dye industry it is used as a wood stain and as a dyeing agent for fur and feathers. A major process for the preparation of *p*-aminophenol is *via* the



hydrogenation of nitrobenzene in the presence of strong acids.¹⁵² Mesoporous carbon nitrides (m-CN) have pendant amino groups¹⁵³ intrinsically present on their mesoporous walls, which can serve as nucleation centers for the grafting of metal NPs, thereby leading to m-CN-material-based catalysts containing well-dispersed catalytically active sites.¹⁵⁴ Very recently, Zuo *et al.* reported the preparation of a N-doped mesoporous carbon supported Ni nanoparticle catalyst (Ni/m-CN) *via* carbonization of Ni-MOF [Ni(Hbtc)(4,4'-bipyridine)] at 700 °C.¹⁵⁵ When the resulting Ni/m-CN catalyst was tested in the catalytic reduction of 4-nitrophenol using NaBH₄ as a stoichiometric reductant, the reaction proceeded very rapidly. The catalytic performance was preserved without any loss of efficiency after six cycles, highlighting its excellent activity and stability upon reuse besides being easily recovered due to the superparamagnetism of the catalyst. Compared to other catalysts in the literature, the activity of the Ni/m-CN catalyst was approximately the same as that of Ag-doped carbon spheres and Fe₃O₄@SiO₂-Ag catalysts^{156,157} and considerably higher than other Ni and Co based catalysts.^{158,159}

Hydrodesulfurization (HDS) is the oil refinery process for removal of sulfur from sulfur-containing organics present in crude oil fractions. The process typically relies on alumina supported heterogeneous catalysts with molybdenum–cobalt or molybdenum–nickel sulfides as active components.¹⁶⁰ Not long ago, in a more sophisticated version of MOFMS, Larabi *et al.* reported the application of the well-defined MOF [Ni₂(dhtp)] (H₄dhtp = 2,5-dihydroxyterephthalic acid),¹⁶¹ as a precursor to bimetallic nickel containing molybdenum sulfide nanoparticles, for HDS reactions.¹⁶² The active material was prepared by subliming Mo(CO)₆ on activated [Ni₂(dhtp)], which has the capacity of grafting metal carbonyls on the organic linker. The porous, highly dispersed, bimetallic Ni–Mo material was sulfided for 4 h at 350 °C by reaction with H₂S in the presence of 42 atm H₂. The HDS activity of the activated catalyst was measured for the conversion of dibenzothiophene to biphenyl or cyclohexylbenzene at 350 °C in the presence of 38 atm H₂. The MOF-derived sulfided material resulted in an active bulk HDS catalyst, stable over several cycles and yielded almost an order of magnitude improvement in the catalytic activity per molybdenum center when compared to the reference unsupported bulk NiMoO₄ material. Furthermore, hydrodenitrogenation (HDN) and hydrogenation activities were also relatively high for this material.

As in the case of Co and Cu, the reduction potential of Ni²⁺ suggests that pyrolysis of Ni(ii)-MOFs tends to form metallic nickel embedded in carbon at a relatively low temperature (250 °C) according to the Ellingham diagram. Metallic Ni is also a well-known catalyst for graphitization of carbon.¹⁶³ Therefore pyrolysis parameters such as temperature during Ni-MOFMS require special attention in order to control graphite layer formation around Ni nanoparticles, which might lead to internal diffusion limitations or even to total isolation of the active phase. It should not escape our attention that preparation of NiO catalysts from Ni-MOF leads to highly porous structured oxides, comprising another appealing, unexplored and simple

application of Ni-MOFMS. On the other hand, grafting on Ni-MOFs has already given way to superior catalytic materials for hydrotreating processes, thus paving the way for the application of these or similar materials in other catalytic processes. In any case, there is still a long way to go regarding MOFMS for this metal.

(i) Ruthenium (Ru)-MOFMS

Ruthenium is one of the rarest metals on Earth. Although it is found uncombined in nature, it is more commonly found associated with other metals like palladium in minerals like pentlandite and pyroxinite and commercially it is obtained from the wastes of nickel refining. The melting point of Ru is 2334 °C, its boiling point is 4150 °C and the reduction potential of Ru²⁺ is −0.45 V.

Ruthenium is used in catalysts for ammonia and acetic acid production. It is also an excellent catalyst for converting deNO_x to N₂, but the oxide is relatively highly volatile, so unsuited for use in three way catalytic converters (TWCS). Ruthenium catalysts are widely known for dissociating the CO bond easily and for being effective for methanation or Fischer–Tropsch synthesis.¹⁶⁴ By and large, molecular ruthenium catalysts are used to perform selective carbon–carbon bond formation by combination of simple substrates and their tolerance toward functional groups has allowed access to high value, multifunctional molecules. Ru catalysts allow the coupling of functional alkenes or alkynes with a variety of unsaturated molecules such as alkenes, dienes, alkynes and diynes. A large range of electron-rich ruthenium or hydridoruthenium complexes are currently used for the formation of cyclic and polycyclic compounds on reaction with substrates containing several pathways, such as C–H bond activation, the distribution of carbene from diazoalkanes and especially their versatility in making a large variety of ruthenacycle intermediates.¹⁶⁵ New ruthenium catalysts enable carbon–carbon, carbon–hydrogen, and carbon–heteroatom bond formation and cleavage, and are able to provide non-classical activation modes.

Very recently, Chen *et al.* provided a novel concept for MOF mediated synthesis by hybridization of clay minerals with metal–organic frameworks as efficient catalysts for benzene hydrogenation.²³ To this end, they grew a ruthenium MOF [Ru₂^{II,III}(btc)₂X_xY_{1.5-x}] (X, Y = Cl[−] or OH[−]) on montmorillonite (Mt) *via* a solvothermal reaction, fabricating a composite material (Ru-MOF_Mt). In order to obtain stable Ru NCs, the prepared hybrids were annealed in H₂ at 300 °C, thus partially decomposing the frameworks and producing Ru particles that act as active centers for hydrogenation. In the catalytic tests, benzene was proven to be completely converted into cyclohexane within 2.5 h at 160 °C and a H₂ pressure of 6.0 MPa. The turnover frequency (TOF) of benzene reached 3478 h^{−1}, which is higher than that reported for Ru/C or other heterogeneous catalysts.^{166,167} Moreover, the Ru-MOF_Mt catalyst was reused 5 times and no obvious degradation was detected, which demonstrated the good stability of the Ru particles protected by both the partly decomposed btc oligomers and the clay layers.



To sum up, the low availability of Ru makes this metal far expensive. Since in principle MOFMS targets high loadings of highly dispersed metals in supported catalysts, applying this technique in this case may seem quite pointless. Alternatively, diluting Ru-MOFs with other solids comprises a more suitable option for this metal. Both organic and inorganic materials might be used for this purpose but always keeping in mind this will directly affect both the porosity of the carbon matrix and the final state of the active metal phase. In any case, the reduction potential of Ru^{2+} predicts a strong tendency during the MOFMS in inert atmosphere to form metallic Ru.

(j) Titanium (Ti)-MOFMS

Titanium is the second most abundant transition metal after iron, its melting and boiling points being 1668 °C and 3287 °C, respectively. For TiO_2 , the melting point ranges from 1830 to 1855 °C and its boiling point from 2500 to 3000 °C. The reduction potentials of Ti^{3+} and Ti^{2+} to metallic Ti are -1.21 V and -1.63 V , respectively, and in the case of Ti^{3+} to Ti^{2+} is -0.37 V . Probably the best-known applications of Ti in catalysis are as TiCl_3 and TiCl_4 Ziegler–Natta catalysts for the production of polypropylene and as TiO_2 catalytic support for different processes and as a photocatalyst. Indeed, during the most recent decades TiO_2 based photocatalysts have attracted great interest due to their excellent chemical stability, nontoxicity, and low price.¹⁶⁸ Its low cost and simple nature of processing make the sol–gel route the common technique for the synthesis of TiO_2 . Nevertheless, the control of TiO_2 in terms of phase, shape and morphology, and especially the nanoscale synthesis of TiO_2 particles through a simple and controllable method remains a challenge.¹⁶⁹

Among Ti based MOFs, MIL-125 is the first example of a highly porous and crystalline titanium(IV) dicarboxylate with high thermal stability and photochemical properties.¹⁷⁰ Its structure is built up from a pseudo cubic arrangement of octameric wheels, built up from edge- or corner-sharing $\text{TiO}_5(\text{OH})$ octahedra, and terephthalate dianions leading to a three-dimensional periodic array of two types of hybrid cages with accessible pore diameters of 6.13 and 12.55 Å.

TiO_2 is widely used as a photocatalyst for CO_2 reduction;¹⁷¹ however, this material often exhibits low efficiency.¹⁷² Noble metal co-catalyst species favour electron–hole separation and induce interfacial electron transfer or allow visible light absorption. In this regard, the potential of gold nanoparticles (GNPs) has received recently considerable attention.¹⁷³ For instance, Khaletskaya *et al.* targeted gold/titania nanocomposites, GNP/ TiO_2 through the pyrolysis of GNP/ NH_2 -MIL-125 nanocrystals and tested them for the UV light promoted reduction of CO_2 to methane.¹⁶⁹ Preformed GNPs, aminoterephthalic acid and titanium(IV) isopropoxide were dissolved in a mixture of *N,N*-dimethylformamide (DMF) and methanol (MeOH) (DMF:MeOH = 1:1 (v/v)). The mixture was heated to 423 K, and kept at this temperature for 15 h which yielded GNP/ NH_2 -MIL-125. GNP/ NH_2 -MIL-125 was calcined in O_2 at 450 °C for 2 h resulting in the selective rutile-phase formation of GNP/ TiO_2 with the retained size and morphology of the

original MOF crystals which enabled the controlled nanoscale synthesis of TiO_2 in terms of phase, shape, and morphology. In catalytic terms, the aforementioned properties translated into high activity of the GNP/ TiO_2 photocatalyst for CO_2 reduction compared to P-25 and AUROLite (commercial Au/ TiO_2) besides not showing any signs of deactivation in the second reaction run. The authors do however propose further improvement of GNP/ TiO_2 materials by (i) directing toward a more uniform distribution of the GNPs, (ii) further downscaling and tailoring of the primary particle size of TiO_2 , (iii) elucidating the effects of light heteroelement doping (C, N) together with GNP loading and (iv) addressing CO_2 reduction under visible light conditions.

Oxidative desulfurization (ODS) has been extensively explored in the recent literature as a means to attain deep desulfurization of petroleum streams.¹⁷⁴ In ODS, sulfur compounds are oxidized to their corresponding sulfoxides and sulfones which can be more easily extracted due to increased polarity. On this subject, McNamara *et al.* not only addressed the performance of Ti-MOF derived catalysts for the oxidation of dibenzothiophene but also focused on the effect of the porosity of the MOF precursor on the final morphology of the resulting MOF-templated material.¹⁷⁵ As well as the microporous MIL-125(Ti), they prepared a hierarchically microporous/mesoporous analogue of MIL-125(Ti) made up of agglomerations of microporous MOF nanoparticles exhibiting interparticle mesoporosity. The main differences in the synthesis of this mesoporous MOF compared to the microporous analogue are the preparation of a dry precursor powder for use in a vapor-assisted crystallization setup and the addition of cetyltrimethylammonium bromide (CTAB) as a templating agent. Both microporous and micro-porous/mesoporous MOF materials were carbonized in Ar at different temperatures ranging from 600 to 1100 °C at a ramp rate of 10 °C min^{-1} for 6 h and were passivated under 1% O_2 in He for 1 h afterwards. The results showed that altering the MIL-125 morphology to obtain interparticle mesoporosity yielded materials with significantly enhanced mesoporosity when pyrolyzed compared to pyrolysis of the strictly microporous MIL-125 material. Furthermore, pyrolysis of the hierarchically micro-porous/mesoporous MIL-125 yielded a carbonaceous material with smaller Ti nanoparticles compared to pyrolysis of the micro-porous MIL-125, which was attributed to the confined environment of the Ti particles embedded in smaller carbon particles. In addition, increasing the pyrolysis temperature led to higher Ti content (>40 wt%) in the final materials and reduced Ti oxide phases including a Ti oxycarbide phase. Oxidation of sulfur-containing dibenzothiophene using *tert*-butyl hydroperoxide revealed that the meso-materials exhibited enhanced activities over the micro-materials pyrolyzed at the same temperature due to their increased mesoporosity and smaller Ti nanoparticle sizes. The above results demonstrated that, by effectively tuning the morphology of the precursor MOFs, the properties of the pyrolyzed materials could be tailored to suit specific applications.

Presumably, the most outstanding feature of Ti-MOFMS is the strong temperature dependence on the carbothermal



reduction of Ti species under pyrolysis conditions which prompts the generation of oxides, carbides, and oxycarbides supported on nanoporous carbon.¹⁷⁶ Thus, as well as the metal loading and the size of the nanoparticles, the carbonization temperature used during the pyrolysis allows the final phase of Ti to be tuned, from Ti^{4+} species to more reduced oxidation states of Ti (Ti^{2+} and Ti^{3+}) *via* carbonization and carbothermal reduction. Otherwise, the synthesis of pure TiO_2 needs to be conducted through calcination in air above 300 °C in the case of $NH_2\text{-MIL-125}$ and 400 °C for MIL-125.¹⁷⁷

(k) Vanadium (V)-MOFMS

Vanadium is the 21st most abundant element in the Earth's crust and the 2nd most abundant transition metal in sea water.¹⁷⁸ The melting point of V is 1910 °C, its boiling point being 3407 °C. Vanadium oxide standards include V_2O_5 , VO_2 , V_2O_3 , and VO , where vanadium is present in oxidation states V^{5+} , V^{4+} , V^{3+} , and V^{2+} , respectively,¹⁷⁹ the prevailing valence states of V in nature being V(III), V(IV), and V(V). Furthermore, the V-C system is a typical binary system which has many different stoichiometries. V_2C , V_4C_3 , V_6C_5 , V_8C_7 , and VC have been synthesized and investigated for many years.¹⁸⁰ Boiling points of V_2O_5 and VC are 2030 and 3900 °C, respectively. The reduction potentials of V^{3+} to V^{2+} and V^{2+} to V are -0.26 and -1.18 V, respectively.

Vanadium oxide or supported vanadium oxide catalysts which contain a vanadium oxide phase deposited on a high surface area oxide support (*e.g.*, Al_2O_3 , SiO_2 , TiO_2) have found extensive applications as oxidation catalysts in the chemical, petroleum and environmental industries. These reactions include the oxidation of sulfur dioxide to sulfur trioxide in the production of sulfuric acid, the oxidative dehydrogenation of propane to propene, the selective oxidation of methane to formaldehyde, the oxidative dehydrogenation of alkanes and the selective photo-oxidation of alcohols.^{181,182} Moreover, transition metal carbides, pure or modified by oxygen, have been extensively studied as catalysts, due to their attractive advantages in terms of activity, selectivity, stability, and resistance to poisoning in hydrocarbon reactions.¹⁸³ Among these, vanadium carbide is of importance for industrial applications due to its excellent high temperature strength, and high chemical and thermal stability even at high temperatures, that make it more resistant to sintering and attrition than other solids.¹⁸⁴

MIL-47 is a vanadium benzenedicarboxylate MOF with chemical formula $V^{IV}O(O_2C\text{-}C_6H_4\text{-}CO_2)$, composed of $V^{IV}O_6$ octahedra linked by a 1,4-benzenedicarboxylate group forming a three-dimensional, orthorhombic, porous solid and it degrades below 400 °C under both nitrogen and air flow.¹⁸⁵

Substantial leaching of active V species often limits the reusability of V-based solid catalysts in liquid-phase oxidation reactions thus calling for the development of more stable, novel materials. Kim *et al.* worked on the synthesis of active vanadium oxide and carbide species dispersed on a carbon support from a V-based MOF template and evaluated its catalytic performance in the liquid-phase oxidation of dibenzothiophene (DBT) with *tert*-butyl hydroperoxide (TBHP).¹⁸⁶

The activated MIL-47(V) was carbonized under Ar with a ramp rate of 10 °C min⁻¹ at temperatures ranging from 600 °C to 1100 °C for 6 h. After cooling to 25 °C under Ar, the resulting catalyst was passivated under 1% of O_2 in He for 1.5 h prior to exposure to ambient atmosphere. Notably, the pyrolysis of MIL-47(V) provided carbon supports with high surface areas ($\sim 300\text{-}400\text{ m}^2\text{ g}^{-1}$), high mesoporosity ($V_{\text{meso}}/V_{\text{pore}} \sim 0.9$), high V loadings (35–70 wt%), and small (~ 18 nm) V crystallites dispersed on the surface. The effect of the pyrolysis temperature was clear-cut on different surfaces and bulk phases of V achieved, higher temperatures leading to increased amounts of V carbide at the surface, even reaching 100% V carbide composition at the surface of the sample carbonized at 1100 °C. The MOF derived V carbide catalysts carbonized at high temperature (1000 and 1100 °C) exhibited good activities and enhanced stabilities compared to the conventional V on activated carbon synthesized *via* a conventional impregnation and to MOF derived V catalysts carbonized at lower temperatures (600 °C). The authors claimed this may likely be attributed to the advantages of the V carbide over the oxide counterparts, which can provide enhanced chemical stability and enhanced anchoring on the carbon support during this reaction and which could not be obtained for the conventional V supported by impregnation on activated carbon even when carbonized at 1100 °C.

Traditionally, vanadium carbide has been prepared by various high temperature reactions. Recently, other synthesis methods have been developed including mechanical alloying, temperature programmed reaction and gas reduction–carburization, aluminothermic reduction of vanadium oxide and carburization of vanadium oxide with an organic reagent such as cyanamide.¹⁸⁴ However, industrial applications of the methods are still limited due to agglomeration problems, wide size distributions, low yields, complex monitoring and high costs.¹⁸⁷ Based on these terms, MOFMS comprises a novel route to obtain transition metal carbides by one step pyrolysis above 400 °C when MIL-47 is being used as the parent MOF.

(l) Zinc (Zn)-MOFMS

Its high availability, low toxicity and environmental benignity¹⁸⁸ make Zn a highly attractive metal for application in catalysis. Zn has a melting point of 419.5 °C and a boiling point of 907 °C. The reduction potential of the zinc ion is -0.76 V and the melting point of its oxide, ZnO , is 2100 °C, its Tamman temperature being 900 °C.¹⁸⁹ Due to its electronic properties, ZnO is mainly used as a photocatalyst and also as a promoter/support in heterogeneous catalysis.

Among Zn-MOFs, MOF-5 and ZIF-8 are the most obvious candidates to be used as templates in MOFMS. Consisting of Zn_2O building blocks linked together by terephthalate bridges to form a zeolite-like, cubic framework, the group of O. M. Yaghi published the structure of MOF-5 in late 1999.¹⁹⁰ Since then, MOF-5 has been subjected to numerous studies, especially in gas storage^{191–194} and separation¹⁹⁵ as well as in catalysis.^{196–198} ZIF-8 ($Zn(\text{mIM})_2$, mIM = 2-methylimidazolate) exhibits a SOD topology comprised of cages connected through six-membered

windows, and is currently the most widely investigated zeolitic imidazolate framework (ZIF) material for a wide range of applications.¹⁹⁹ The unique properties of ZIF-8 such as pressure-dependent pore size, reversible guest trapping and amorphization²⁰⁰ have boosted its use for gas separation,²⁰¹ sensing²⁰² and catalysis.²⁰³

Depending upon the heat treatment Zn-MOFs give way to two different MOF derived catalytic materials: (i) nanostructured metal oxides and (ii) hierarchical nanoporous carbons.

In order to prepare nanostructured ZnO from Zn-MOFs, the typical approach consists of calcining the MOF under air at a temperature above the decomposition temperature of the framework, around 300 °C for both MOF-5²⁰⁴ and ZIF-8.²⁰⁵ However, as the reduction potential of Zn²⁺ is lower than -0.27 V and thus below 900 °C zinc ions are unable to reduce to a zero oxidation state, the calcination under inert atmosphere of Zn-MOFs also gives the corresponding metal oxide. The main advantage of MOFMS compared to other preparation methods lies in the high porosity and structure/morphology control of the resulting MOF derived ZnO nanoparticles²⁰⁶ which results in improved catalytic performance.

The major limitation for achieving high photocatalytic efficiency in the ZnO nanostructure systems is the fast recombination of charge carriers.²⁰⁷ Therefore, the design and modification of ZnO photocatalysts with high sensitivity and reactivity has attracted much attention.²⁰⁸ In this field, Yang *et al.* demonstrated that ZnO nanoparticles prepared by calcination of MOF-5 in air at 600 °C, formed hierarchical aggregates with a three-dimensional cubic morphology. These aggregates exhibited high photocatalytic degradation of Rhodamine-B (RhB) under UV irradiation, comparable to degradation by the commercial P-25 TiO₂ catalyst.²⁰⁹ The authors investigated the outstanding photocatalytic activity by measuring the room temperature photoluminescence (PL) of the sample. The PL spectrum indicated that the rate of recombination between the photogenerated holes and electrons was lowered on the surface of the prepared ZnO crystals thus resulting in a superior photocatalytic activity. In other work, Cao *et al.* developed a simple method to prepare metal oxide/3D graphene network (3DGN) composites using ZIF-8 as the precursor of the metal oxide and 3DGN as the backbone. After the preparation of the ZIF-8/3DGN composite, the synthesis consisted of a facile two-step annealing process under Ar at 450 °C for 1 h, which conferred the ZIF-8/3DGN material with a more textured and rougher surface, followed by calcination in air at 380 °C for 1 h to give the final ZnO/3DGN material.²⁴ In their case, the improved performance of the ZnO/3DGN composite in the photodegradation of the dye methylene blue (MB) under irradiation with UV light was ascribed to the high specific surface area of the composite whereas the prevention of the recombination of photoinduced holes and electrons was attributed to good contact and effective interaction between 3DGN and ZnO. Similarly to Cao *et al.*, Pan *et al.* also selected nonmetal doping in order to improve the low efficiency of charge separation in ZnO nanoparticles, using in this case the organic linker of the pristine MOF as the carbon source.²¹⁰ The authors prepared C-doped ZnO *via* two-step

pyrolysis under N₂ of ZIF-8 and applied it in the photodegradation of RhB and phenol and photo electrochemical (PEC) water splitting. To this end, ZIF-8 was first calcined in air at 350 °C for 2 h (at a heating rate of 5 °C min⁻¹) to partially remove the organic groups resulting in a porous structure, then the temperature was increased to 400 °C for 1 h (at a heating rate of 2 °C min⁻¹) to remove the organic residues and increase the crystallization of ZnO. Compared to a commercial C-doped ZnO catalyst, the results showed that the ZIF-8 derived catalyst exhibited around 2-fold higher photoactivity in both the photodegradation of RhB and phenol and 4-fold higher in PEC water splitting. The authors claimed that the inter-connected porous structure of ZnO nanoparticles played an important role in mass transfer, charge transfer and light absorption during the photoreactions.²¹¹

Besides its use in photocatalysis, MOF derived ZnO is also advantageous in some other catalytic applications. During the catalytic oxidation of CO, creating a high density of noble metal nanoparticles and a high support interface is the key to maximize catalytic performance.^{212–214} For this purpose, Liu *et al.* employed MOF-5 as a precursor to prepare ZnO-supported Pt nanoparticles which exhibited excellent catalytic activity for CO oxidation.²¹⁵ Nanocrystalline ZnO-supported platinum nanoparticles were synthesized by introducing inorganic platinum salt into the pores of MOF-5 followed by heating in air at 600 °C for 1 h, which allowed the platinum nanoparticle size to be tuned by changing the concentration of the platinum precursor. The Pt/ZnO obtained from the MOF resulted in a catalytic activity twice as high as that of Pt/ZnO obtained by conventional methods. The authors suggested that the good catalytic activity of the Pt/ZnO samples may result from metal (Pt)-support (ZnO) interaction and depended on the Pt particle size.

Moving forward in this topic, the removal of Zn from the framework during MOFMS results in highly porous hierarchical carbons. As shown in the Ellingham diagram, above 950 °C the Gibbs free energy of formation for C is lower than that for ZnO meaning that carbon is able to reduce zinc oxide, the carbon itself being oxidized to CO, and to CO₂ above 1050 °C. Accordingly, in the vast majority of carbons from Zn-MOFs the pyrolysis of the pristine MOF takes place around 900–950 °C for several hours in order to completely evaporate the metal. An alternative and far less employed approach consists of pyrolysis at lower temperature decomposing the framework to form the carbon, and subsequently removing the ZnO by acid leaching.

These MOF mediated porous carbons find their main application in electrocatalysis. Two main strategies exist in order to optimize the design of ORR catalysts thus boosting their performance, which consist of (i) enhancing Pt dispersion in the catalyst and/or (ii) replacing the expensive Pt with non-precious metals. In order to maintain good Pt nanoparticle dispersion, recent efforts have been focused on the development of new carbon support nanostructures in an attempt to further improve the activity and stability of PEMFCs.²¹⁶ In their work, Khan *et al.* developed highly efficient ORR electrocatalysts based on Pt/Ni bimetallic nanoparticles dispersed on highly porous carbon obtained *via* pyrolysis of MOF-5.²¹⁷



Firstly, MOF-5 was heated in an argon atmosphere at 950 °C for 9 h for improved graphitization and removal of all impurities including Zn/ZnO. Subsequently, the obtained carbon material (PC 950) was sonicated in ethylene glycol, heated up to 100 °C and H₂PtCl₆·6H₂O and Ni(NO₃)₂·6H₂O in ethylene glycol were separately added dropwise. Finally, the resultant mixture was heated up to 180 °C for 4 h to complete the polyol reduction synthesis method.²¹⁸ As a result, Pt-Ni/PC 950 (15:15%) demonstrated superior ORR activity over a Pt/C (20%) commercial catalyst in terms of both E_{onset} and $E_{1/2}$, attributed to the synergistic effect between Pt–Ni nanoparticles and the carbon encapsulating effect of the MOF derived material. Previously these authors applied MOF-5 as a template for the preparation of porous carbon as an electrode material in ethanol fuel cells (DEFCs).²¹⁹ In this case, MOF-5 was carbonized at 900 °C in Ar for 6 h and the carbon material (PC-900) was deposited with PtFe (20%) nanoparticles using the same polyol reduction method. The electrocatalytic capability of the catalyst for ethanol electro-oxidation was investigated and compared with those of PtFe and Pt supported on Vulcan XC72 carbon catalysts (PFe/CX-72 and Pt/XC-72) prepared *via* the same method. They observed that the catalyst PtFe/PC-900 showed an outstanding normalized activity per gram of Pt and superior power density compared to the commercially available carbon-supported catalysts.

The second course of action involves reducing the cost and promoting a wider use of PEMFCs replacing the expensive Pt based electrocatalysts with a lower-cost alternative.^{220,221} Fe-Based cathode catalysts are promising contenders, but their power density is low compared with Pt-based cathodes, largely due to poor mass-transport properties. In view of this, Proietti *et al.* used ZIF-8 as a host for Fe and N precursors by means of mixing iron acetate and phenantroline together with ZIF-8 using low-energy ballmilling.¹²³ The mixture was pyrolyzed in Ar at increasing temperatures and optionally in NH₃ at 950 °C, creating highly porous Fe/N/C composites. With an increase in the Ar pyrolysis temperature to 1050 °C, Zn could be removed completely to get purely microporous carbons. NH₃ pyrolysis resulted in an additional 17 wt% loss by hydrogasification, doubling the BET area and introducing substantial mesoporosity besides increasing the nitrogen content in the sample. The introduction of mesopores together with the general increase in the microporous surface area and the nitrogen content resulted in an over four times increased ORR activity compared to the microporous carbon.

In 2009, a new class of catalysts based on earth-abundant carbon materials was discovered as an efficient, low-cost, metal-free alternative to platinum for the ORR in fuel cells. Since then, tremendous progress has been made, and carbon-based metal-free catalysts have been demonstrated to be effective for an increasing number of catalytic processes.²²² A good example of this is work done by Aijaz *et al.* who designed high-surface-area N-decorated nanoporous carbons using ZIF-8 along with furfuryl alcohol (FA) and NH₄OH as the secondary carbon and nitrogen sources, respectively.²²³ FA and NH₄OH were first introduced into the cavities of activated ZIF-8 at room temperature. After filtration and washing with ethanol, the FA–NH₄OH/ZIF-8 composite was

heated in Ar at 80 °C for 24 h, then at 150 °C for 7 h for FA polymerization, and finally calcined at 600, 700, 800, 900, or 1000 °C for 8 h. Among the MOF derived electrocatalysts, the one calcined at 900 °C showed the highest onset potential of 0.83 V, compared with 0.95 V for the commercial Pt/C catalyst according to cyclic voltammetry measurements. The high ORR catalytic activity was associated with the large number of active basic N sites and the high surface area, which provided proper channels with easy mass diffusion.

Along a similar line, research by Zhang *et al.* showed that the introduction of ordered mesopores by using Te nanowires (Te-NWs) as a templating agent dramatically increased the performance of cathode ORR in alkaline media (Fig. 6).²²⁴ Te-NWs were added to the ZIF-8 synthesis mixture, and after crystal growth on the NW surface, the composite was pyrolyzed at 1000 °C to vaporize both the Te and the Zn phase. Comparison of the microporous ZIF-8 derived carbon with the Te-NW MOF-templated mesoporous carbon showed large differences in performance. The ORR activity of the templated carbon was much higher and close to that of commercial Pt-catalysts. The hollow carbon nanofibers were subsequently P-doped to create more ORR active sites, and the resulting catalyst showed the best catalytic activity, outperforming the commercial 20 wt% Pt/C.

Apart from their application in electrocatalysis, nanostructured carbons from Zn-MOFs find their application in other fields of catalysis.^{21,225} For instance, Feng *et al.* prepared nanoporous carbons by carbonization of MOF-5 in a N₂ atmosphere at different temperatures, followed by the introduction of silver and palladium precursors using incipient wetness impregnation.²²⁵ Ag₃Pd₁₂/NPCs obtained through pyrolysis at 900 °C exhibited high activity and selectivity in dehydrogenation of formic acid in sodium formate under ambient conditions in comparison with reported heterogeneous catalysts for this reaction. The authors ascribed the enhanced catalytic performance of Ag₃Pd₁₂/NPCs to its Pd⁰, Ag⁰ and AgPd alloy composition as well as to the small particle size and good dispersion of AgPd nanoparticles on the carbon support.

The analysis here is clear. Both the intriguing properties of ZnO for photocatalysis together with the low boiling point of Zn have certainly boosted research on the MOFMS of Zn-MOFs in order to afford nanostructured ZnO and hierarchical carbons for their catalytic application. The simplicity of the technique for the preparation of both catalytic materials makes Zn-MOFMS up-and-coming for its future industrial implementation. In the case of ZnO, the technique implies pyrolysis in an inert atmosphere, calcination in air or a combination of both at temperatures that range from 400–600 °C. Although the technique has been largely applied in the field of photocatalysis, shifting its application to other catalytic processes will require greater effort with regard to both preparation of the materials and catalytic testing. Regarding the preparation of carbon supports from Zn-MOFs by pyrolysis at 900–1000 °C, the MOFMS technique has been mainly applied in electrocatalysis for supporting precious and non-precious metal nanoparticles and also for electrocatalytically active metal-free carbons. However, the number of possibilities seem to be huge in this field which



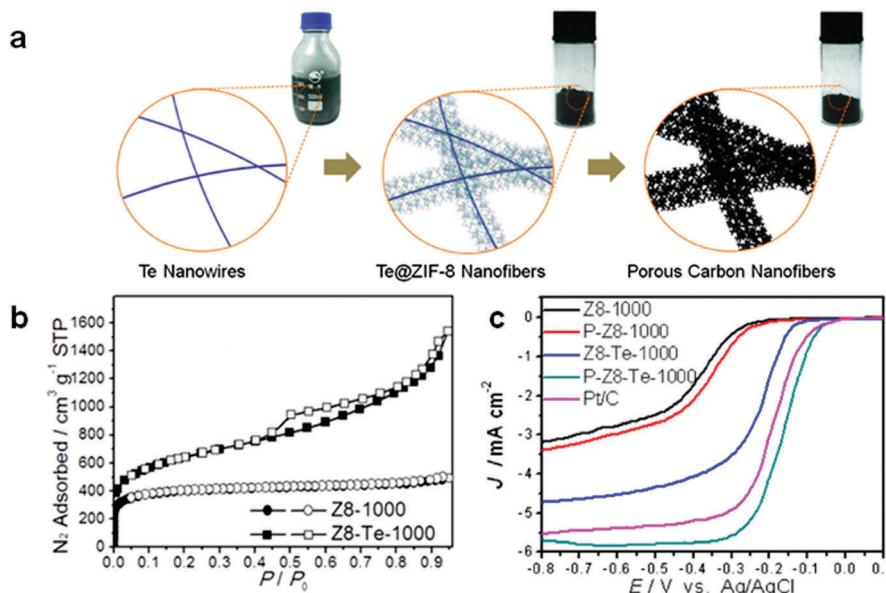


Fig. 6 (a) Illustration of the nanowire-directed templated synthesis of ZIF-8 nanofibers and derived porous doped carbon nanofibers; (b) N_2 sorption isotherms (@77 K) of Z8-Te-1000 and Z8-1000; (c) linear sweep voltammetry (LSV) curves of all catalysts in O_2 -saturated 0.1 M KOH with a sweep rate of 10 mV s⁻¹ and electrode rotation speed of 1600 rpm. Reprinted with permission from Zhang *et al.*, 2014.²²⁴ Copyright 2016, American Chemical Society.

allows metal-free carbons with improved textural properties to be obtained by simple pyrolysis of the parental Zn-MOF.

(m) Zirconium (Zr)-MOFs

Zirconium occurs in about 30 mineral species, the major ones being zircon and baddeleyite. Zirconium metal is produced commercially by first converting zircon to zirconium chloride, and then reducing the chloride with magnesium. The melting point of Zr is 1855 °C and its boiling point is 4409 °C. The most common oxidation state of zirconium is IV and the reduction potential of Zr^{4+} is -1.54 V.

Zirconia is a very important ceramic material with growing applications in heterogeneous catalysis both as an active material and a support. As a support, ZrO_2 has attracted considerable interest in a variety of catalytic systems since it is characterized to be the only one metal oxide that can possess four different chemical properties on the surface: acidic, basic, oxidizing and reducing properties.²²⁶ Accordingly, it has been extensively used in catalysis, for example supporting Au, Ag, and Cu catalysts in the water–gas shift reaction, methanol synthesis from CO_2 and H_2 , CO oxidation and in the form of nanostructures, in photocatalytic reactions like CO_2 reduction.²²⁷ Besides, decomposition of NO has been suggested to occur in Lewis acid sites of zirconia nanoparticles,²²⁸ synthesis of aromatic aldehydes by the hydrogenation of carboxylic acids with a ZrO_2 catalyst has been already established²²⁹ and various authors have reported on the catalytic activity of zirconia for the selective dehydration of secondary alcohols to terminal alkenes as well as for the hydrogenation of CO to isobutene.^{230–232}

The strength of Zr^{IV} –O bonds has been exploited to develop highly stable metal–organic frameworks (MOFs) that are tolerant to extreme conditions, including chemical functionalization and

catalytic reactions and they possess high mechanical strength and high thermal and chemical stability (up to ~500 °C, pH 1–10).²³³ UiO-66 is composed of $[Zr_6(\mu_3-O)_4(\mu_3-OH)_4]^{12+}$ nodes, each linked to 12 carboxylates of terephthalate ligands to form tetrahedral and octahedral cages; however, there are many defects from missing linkers in this MOF, with the resulting otherwise open metal sites occupied by hydroxo ligands. The node topology of the more highly porous²³³ NU-1000 has been determined to be $[Zr_6(\mu_3-O)_4(\mu_3-OH)_4(OH)_4(OH_2)_4]^{8+}$ and it is coordinated to eight tetratopic linkers of 1,3,6,8-tetrakis-(*p*-benzoic-acid)pyrene (H_4TBAP_y) to form triangular and hexagonal pores.²³⁴

Very recently Malonzo *et al.* demonstrated the effectiveness of nanocasting of NU-1000 with silica for the preparation of highly thermally stable MOF derived single site catalytic clusters for high temperature reactions (Fig. 7).²⁵ NU-1000 is an excellent Lewis acid catalyst owing to the high concentration of the oxozirconium clusters in its structure, which are easily accessible through its mesoporous channels. By condensing tetramethylorthosilicate within the NU-1000 pores *via* vapour phase HCl treatment, they created a silica layer on the inner walls of the MOF which was subsequently calcined in air at 500 °C for 1 h, giving place to the $Zr_6@SiO_2$ composite. The silica layer provided anchoring sites for the oxozirconium clusters in NU-1000 after the organic linkers were removed at high temperatures whereas the Lewis acidity was enhanced by dehydration of the clusters during the heat treatment at 500 °C.²³⁵ FT-IR spectroscopy of pyridine adsorption showed that it contained appreciable amounts of Brønsted and Lewis acid sites, and both sites remained accessible to pyridine molecules. Besides, the catalytic activity of $Zr_6@SiO_2$ was demonstrated through the isomerization of glucose to fructose and

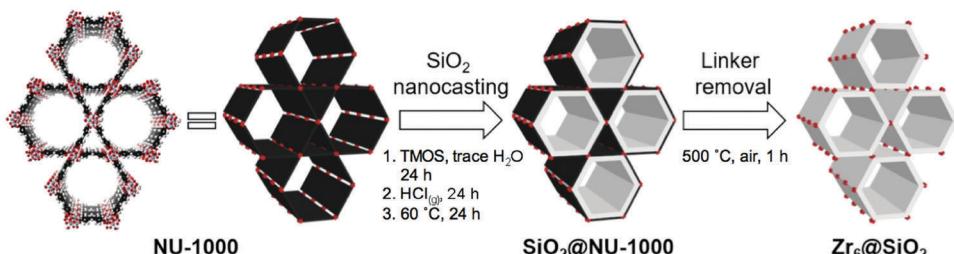


Fig. 7 Stabilization of the oxozirconium clusters in NU-1000 by nanocasting with silica. The left image represents the NU-1000 structure with cluster nodes (red) and linkers (black). The white layer in the middle image represents silica. Reprinted with permission from ref. 25. Copyright 2016, American Chemical Society.

compared to that of NU-1000, calcined NU-1000 and control samples of oxozirconium clusters in non-templated silica supports. The authors concluded that the ability to retain catalytic activity, compared to other materials, through site isolation and the templated porosity despite being heated to high temperatures makes nanocasting a useful technique for applying MOF-based catalytic clusters for catalytic processes that may require high temperatures.

Tungstated zirconia WO_x/ZrO_2 (WZ) is receiving increasing attention as a solid superacid in petrochemistry including alkane's isomerization,²³⁶ dehydration of alcohols²³⁷ and so on. Generally, the tetragonal phase of zirconia (t-ZrO_2) can be generated with the aid of tungstate species after calcination at $600\text{--}900^\circ\text{C}$ forming WO_x species and WO_3 microcrystals on the surface of zirconia. The existence of hydroxyl groups and missing linker defects make the surface structures of zirconium octahedrons in UiO-66 similar to those in amorphous zirconium hydroxide, the commonly used supports in the synthesis of WZ.²³⁸ In this context, Wang *et al.* addressed the synthesis of tungstate immobilized UiO-66-derived WZ as strong solid acids toward high catalytic performance for acetalization of benzaldehyde.²³⁹ Ammonium meta-tungstate was first immobilized in the pores of UiO-66 by the double-solvent technology utilizing water and hexane and the as-prepared $\text{WO}_x@\text{UiO-66}$ was heated at different temperatures (550, 650, 750 and 850 °C) in air for 6 h. Among all the catalysts, UiO-2-650 (where 2 refers to the adsorbed amount of ammonium meta-tungstate and 650 to the calcination temperature) led to the highest conversion (86.0%) toward benzaldehyde's acetalization in comparison with the 44.9% conversion obtained under the same conditions with the traditional zirconium hydroxide-prepared WZ, besides retaining its original activity for six cycles. The authors concluded that at moderate tungsten contents and calcination temperature, the MOF-derived WZ had various acid sites with different acid strength, and owned the highest amounts of acids due to highly concentrated amorphous polytungsten species. As a consequence, these solid acids of WZ exhibited enhanced catalytic activity compared to the traditional WZ prepared from precipitated zirconium hydroxide.

The development of dual-heteroatom-doped nano-porous carbon electrocatalysts for the ORR is particularly attractive as a feasible solution for the commercialization of fuel-cell technology. Heteroatom-doping approaches with N^{240} and P^{241}

among others have been employed widely to improve the electrocatalytic performances of target non-precious-metal catalysts. On the other hand, amine-tagged MOFs can be used to graft functional groups.²⁴² In view of these facts, Fu *et al.* took advantage of post-synthetic modification (PSM) and one-step pyrolysis to produce UiO-66-NH₂-derived phosphorus-nitrogen co-doped porous carbon materials (P-N-carbons) from (Zr) UiO-66-NH₂ and used them as electrocatalysts in the ORR.²⁴³ The as-synthesized UiO-66-NH₂ was added to aqueous solutions of the phosphorus source glyposine, which reacts with the -NH₂ groups, to obtain the UiO-66-NH₂-based P-N-rich precursors through vigorous stirring for several hours at 50°C and then the P-N-rich precursor was washed with excess water to remove unreacted glyposine. Since the UiO-66-NH₂ framework possesses high thermal stability up to 500°C , the P-N-rich precursor was pyrolyzed at a preset temperature ($T = 600, 800\text{--}900, 950$, and 1000°C) in Ar for 3 h and finally soaked with HF solution to remove Zr and its derivatives thus leading to P-N-rich porous carbons. Among all the prepared samples, P-N-carbon-950 exhibited the best electrocatalytic activity similar to that of commercial 20% Pt/C and also better stability and increased tolerance to the methanol crossover effects. Characterization indicated that both the high electrical conductivity and the N and P content of the prepared P-N-carbon-950 played a key role in its electrocatalytic activity for the ORR, because the two factors significantly affect the electron transportation in electrodes and active sites for oxygen reduction.

In the case of Zr-MOFMS, calcination in air is deemed the optimal technique to produce Zr oxides that can act both as support or active phase, and are especially promising in acid catalysis. Grafting on UiO-66 comprises another possibility to incorporate different functionalities on the MOF and therefore on the resulting MOF derived catalyst.

(n) Bimetallic MOFMS

This last section does not deal with a particular metal but with bimetallic MOF derived catalysts. Bimetallic catalysts may open the door to solids with enhanced selectivity, activity, and stability, since they often show electronic and chemical properties that are distinct from those of their parent metals due to "synergistic" effects between the two metals. Bimetallic catalysts started to gain considerable commercial interest in the 1960s for their use in hydrocarbon reforming and since then



have inspired extensive investigations on their possible applications.²⁴⁴

Hetero-bimetallic MOFs have gained particular attention as a subclass of MOFs because of their designable framework structures that are modularly built from two transition-metal clusters as nodes and organic ligands as struts.²⁴⁵ Hetero-bimetallic MOFs not only combine the advantages of homogeneously located metal species and tuned concentrations of different metals in the frameworks, but they also expand the metal range to construct various pseudo-mixed-component MOFs bearing “on-demand” functionalities and applications, as the metals are not restricted to the same valence and coordination geometry.²⁴⁶ All this boosts the application of hetero-bimetallic MOFs as precursors to prepare metal/mesoporous carbon composite catalysts for electro-, photo and also conventional heterogenous catalysis.

It is remarkable, however, that the vast majority of bimetallic MOF mediated synthesis reactions reported so far employ Zn as a sacrificial template in bimetallic MOFs in order to gain further control of the porosity of the resulting monometallic material or sometimes bimetallic but after impregnation of an additional metal not in the framework of the parent MOF. You *et al.* reported a bimetal-MOF (bi-MOF) self-adjusted synthesis of support-free porous Co–N–C nanopolyhedron electrocatalysts by pyrolysis of a Zn/Co bi-MOF without any post-treatments.²⁴⁷ Different mixtures of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ were dissolved in methanol and subsequently poured into methanol containing 2-methylimidazole (MeIM). After thorough mixing and incubation at room temperature, the as-obtained precipitates were centrifuged and washed with ethanol several times, resulting in the corresponding $\text{Zn}_x\text{Co}_{1-x}(\text{MeIM})_2$ bi-MOFs. Finally, the corresponding bi-MOF was heated to the desired temperature (800, 900, and 1000 °C) for 3 h under Ar to obtain the self-supported Co–N–C-*x*-*y*, where *x* represents the molar content of Zn ions in the initial bi-MOF precursor and *y* represents the carbonization temperature. Electrocatalytic studies showed that the carbonization at 900 °C of $\text{Zn}_{0.8}\text{Co}_{0.2}(\text{MeIM})_2$ resulted in the most active Co–N–C for the ORR with a half-wave potential more positive than that of commercial 20 wt% Pt/C and a kinetic current density 3.1 times that of Pt/C in 0.1 M KOH, and excellent stability and methanol tolerance. It also exhibited ORR activity comparable to and stability much higher than those of Pt/C in acidic and neutral electrolytes. The authors argue that the presence of initial Zn forms a spatial isolation of Co that suppresses its sintering during pyrolysis, and Zn evaporation also promotes the surface area of the resultant catalysts besides its composition, and morphology, and hence the ORR activity of Co–N–C could be tuned by the Zn/Co ratio. Various characterization techniques revealed that the superior activity and strong stability of Co–N–C originated from the intense interaction between Co and N, the high content of ORR active pyridinic and pyrrolic N, and the large specific surface area.

Similarly, Wang *et al.* reported an analogous and highly efficient Co–N–C catalyst with highly dispersed Co sites derived from a ZnCo bimetallic ZIF and successfully tested it in the ORR.²⁴⁸ In this case the authors prepared ZnxCo nanocrystals

(where *x* is defined as the molar percentage of Co/(Co + Zn) in the starting materials) by a surfactant mediated growth method and the resultant bimetallic ZIF was calcined at higher temperature. Typically, a mixture of surfactants and 2-methylimidazole were dissolved in deionized water under sonication to give a homogeneous emulsion and then the zinc and cobalt mixed metal nitrate solution was poured into the emulsion. The solution was stirred at 60 °C for 4 h to give a turbid suspension and the ZIF–(100 – *x*)ZnxCo nanocrystals were separated by centrifugation, washed three times with deionized water and two times with ethanol and then dried at 80 °C under vacuum for about 12 h. Finally, the ORR catalysts were prepared by pyrolysis of the ZIF–(100 – *x*)ZnxCo nanocrystals in Ar at 1000 °C at 5 °C min⁻¹ for 60 min and allowed to cool down to room temperature. In this case, the authors show that the best ORR performance is achieved when 5% Zn is substituted by Co in the MOF precursor. The resulting C–95Zn5Co catalyst outperformed the commercial Pt/C catalyst with 20 wt% Pt loading in terms of half-wave potential. The authors suggested that the homogeneous distribution of Zn and Co in the bimetallic ZIF was critical for achieving high dispersion of Co in the derived catalyst, which was facilitated by both the isomorphous nature of ZIF-67 and ZIF-8 and the surfactant mediated growth method.

For the ORR, a core–shell structure may improve the performance and concurrently increase methanol tolerance and stability of the catalysts⁶⁵ and for the HER, the core–shell structure is anticipated to not only enhance the corrosion resistance of the metals in acidic electrolytes but also improve the HER activity *via* the synergistic contributions from both the core and shell.²⁴⁹ No sooner said than done, Lu *et al.* prepared core–shell nanocomposites based on Au nanoparticle@zinc–iron embedded porous carbons (Au@Zn–Fe–C) derived from bifunctional MOFs as electrocatalysts for the ORR and HER.²⁵⁰ Experimentally, Au nanoparticles were prepared by reducing HAuCl_4 at 140 °C with DMF both as a solvent and as a reducing reagent in the presence of poly(vinylpyrrolidone) (PVP) which stabilized the Au nanoparticles.²⁵¹ Then, a Zn–Fe-MOF shell was grown on the Au nanoparticle surface by using iron(III) acetylacetone ($\text{Fe}(\text{acac})_3$) as the iron source, $\text{Zn}(\text{NO}_3)_2$ as the zinc source, H₂bdc as the organic linker, and a DMF–ethanol mixed solution as the solvent. PVP was also added in the above mixture as a stabilizing reagent for the formation of a uniform Zn–Fe-MOF shell. Finally, the Zn–Fe-MOF shell was converted to porous carbons embedded with zinc–iron compounds by calcination in Ar at 600 °C for 2 h at a heating rate of 2 °C min⁻¹, forming core–shell Au@Zn–Fe–C hybrids. As a result, single Au nanoparticles of 50–100 nm diameter were encapsulated within porous carbon shells embedded with Zn–Fe compounds. The resulting Au@Zn–Fe–C hybrids exhibited excellent electrocatalytic performance for both the ORR and HER. The encapsulated Au nanoparticles played an important role in determining the electrocatalytic activity for the ORR and HER by promoting electron transfer to the zinc–iron-embedded porous carbon layer, and the electrocatalytic activity was found to vary with both the loading of the gold nanoparticle cores and the thickness of the metal–carbon shells.



Xu *et al.* synthesized a series of carbon composite materials containing metal oxides (ZnO and/or TiO_x) by pyrolyzing a Zn–Ti heterometallic MOF as the precursor and the photocatalytic activities of these composites were evaluated by the photodegradation of MB in aqueous solutions.²⁵² In a previous work, the same authors developed porous TiO_x/C composite materials by pyrolyzing a Ti-containing MIL-125 which exhibited better photocatalytic activities than the benchmark P-25 TiO₂ during the photodegradation of MB.²⁵³ In order to improve the accessibility to TiO_x active sites, the authors addressed the preparation of highly microporous photocatalysts by pyrolyzing a previously reported Zn–Ti heterometallic MOF ZTOF-1.²⁵⁴ The as-prepared ZTOF-1 was held at 80 °C for 45 minutes in Ar and then pyrolyzed under the same Ar atmosphere at different temperatures of 400, 600, 800, and 1000 °C for 5 h affording ZTOF-1-P4, ZTOF-1-P6, ZTOF-1-P8, and ZTOF-1-P10 materials, respectively. ZTOF-1-P10 demonstrated the best catalytic activity, under a UV light source outperforming benchmark P-25 TiO₂. ZTOF-1-P10 exhibited a dramatically increased surface area due to the carbothermal reduction of Zn²⁺ into metallic Zn followed by vaporization as an extra pore-forming mechanism. Because Zn and Ti are adjacent in the heterometallic secondary building units (SBUs) of the MOF precursor, the catalytically active TiO_x sites generated during pyrolysis would remain in the pores and channels formed by Zn vaporization and were readily accessible to MB, leading to a huge increase of photocatalytic activity. According to the authors, the excellent photocatalytic activity of ZTOF-1-P10 lies in (i) the substantial amounts of TiO and Ti₂O₃ (TiO_x) which have been reported to have a better photocatalytic activity than pure TiO₂ because of the higher density of oxygen vacancies for better trapping of electrons, (ii) the highest Ti content among all the pyrolyzed samples and (iii) the dramatically increased surface area and porosity due to the reduction and evaporation of metallic Zn. In addition, the special rearrangement of Zn and Ti in the SBUs of the ZTOF-1 precursor makes TiO_x formed during pyrolysis remain in the pores and channels generated by Zn evaporation thus making TiO_x active sites easily accessible by substrates leading to high reaction kinetics.

Yang *et al.* worked on the preparation of Ni/nanoporous carbon (Ni/NPC) composites by direct pyrolysis of nonporous heterobimetallic zinc–nickel–terephthalate frameworks (Zn_{1-x}Ni_xMOF, $x \approx 0\text{--}1$) for the 4-nitrophenol reduction reaction.²⁴⁵ Mixing different amounts of Zn(NO₃)₂·6H₂O and Ni(CH₃COO)₂·4H₂O with terephthalic acid (TPA) in DMF afforded nonporous even crystals of Zn_{1-x}Ni_xMOF which were carbonized up to 1223 K in N₂ at a rate of 5 K min⁻¹ and maintained for 2 h to yield Ni_x/NPC metal/mesoporous carbon composites with uniform pore sizes and evenly dispersed metal sites. Both the rate constants and the TOF values suggested higher catalytic efficiency of Ni_{1.0}/NPC and Ni_{0.1}/NPC, which were more active than the previously reported NiCo₂ alloy²⁵⁵ and the Ni NPs in a spherical polyelectrolyte brush nanoreactor²⁵⁶ and the values are also comparable to those of soft-templated Ni NCs on mesoporous carbons.²⁵⁷ The Ni_{1.0}/NPC catalyst was facilely separated from the reaction solution by using a magnet and exhibited similar catalytic performance without any visible reduction in the conversion within 20 min

even after recycling over six runs. The authors concluded that their work not only introduces a new class of material with the highest possible precursor site density and compatible textural properties, but also provides a distinct platform with abundant model compounds that can be used to study the correlation between precursor composition and catalytic performance.

Cu/ZnO catalysts are widely used in industrial production, environmental protection and biorefineries, therefore finding a new and efficient method for the controlled synthesis of Cu/ZnO hybrids is of great importance. Recently, Zheng *et al.* have prepared a novel Cu/ZnO catalyst with nanosized ZnO particles dotted on Cu *via* the calcination and reduction of the Cu(Zn)-HKUST-1 precursor.²⁵⁸ ZnO was dispersed in deionized water and sonicated for 10 min and then mixed with DMF. Subsequently Cu(NO₃)₂·3H₂O dissolved in deionized water and benzene-1,3,5-tricarboxylic acid (H₃btc) dissolved in ethanol were consecutively added under vigorously stirring. Then the solid product was immediately filtered, washed with ethanol and dried. The as-synthesized Cu_xZn_{3-x}-HKUST-1 was slowly heated (2 °C min⁻¹) to 400 °C in air for 4 h yielding Cu_xZn_{3-x}O₃. The calcined Cu_xZn_{3-x}O₃ was reduced in 80 mL min⁻¹ H₂ at 350 °C for 1 h before the hydrogenolysis reaction, and these reduced catalysts were identified as Cu_x/ZnO. It was confirmed that all the catalysts synthesized from HKUST-1 were efficient for the hydrogenolysis of glycerol. Moreover the authors found that the Cu_{1.1}/ZnO catalyst derived from Cu_{1.1}Zn_{1.9}(btc)₂·9.4(H₂O) was more active and stable than Cu/ZnO prepared *via* solvent-free grinding and co-precipitation methods and inferred that the interface between Cu and ZnO played a crucial role in its catalytic performance, making ZnO doped Cu more stable than ZnO plate supported Cu particles.

Some other authors made use of bimetallic MOFMS to prepare bimetallic catalysts, retaining and thus taking advantage of both metals in the framework. Kong *et al.* prepared heterometallic carbide nanoparticles (Co₃InC_{0.75}) embedded in nitrogen-enriched carbon by pyrolysis of an indium-MOF with entrapped cobalt dimers for the ORR.²⁵⁹ Co₃InC_{0.75} nanoparticles with a CaTiO₃-typecubic structure embedded within carbon (Co₃InC_{0.75}@C) were prepared by pyrolysis of CPM-16-Co-In, which is a MOF with the cobalt–indium–btc framework. In CPM-16-Co-In, V-shaped Co₂(OH) dimers were grafted onto zeolitic In-btc networks by carboxyl groups, and uniformly arranged in the channels of the framework. By directly heating these crystals at 700 °C for 2 h in Ar the purple rod crystals of CPM-16-Co-In were transformed into Co₃InC_{0.75}@C. To enrich the nitrogen content in Co₃InC_{0.75}@C, melamine was introduced as the nitrogen source. The pyrolysis of the mixture, prepared by grinding CPM-16-Co-In and melamine with a mass ratio of 1:2, at 700 °C for 2 h in Ar, led to the formation of Co₃InC_{0.75} in the CN_x matrix instead of the C matrix. These Co₃InC_{0.75} heterometallic carbide nanoparticles embedded in N-enriched carbon exhibited an impressive ORR activity comparable to the commercial Pt/C (10 wt%) catalyst in alkaline medium and had superior durability and methanol tolerance. The authors proposed that the presence of indium in Co₃InC_{0.75} prevented the decomposition of heterometallic carbides



into metallic cobalt and carbon at high temperature and the carbon derived from the btc-indium framework coated on the formed $\text{Co}_3\text{InC}_{0.75}$ particles prevented them from excessively aggregating leading to high electrocatalytic performance.

Long *et al.* reported a facile and general approach to fabricate bimetallic and trimetallic transition alloy NPs embedded in an N-doped carbon matrix by a direct pyrolysis method and their performance in hydrogenation reactions.²⁶⁰ With this in mind, the authors firstly developed a series of novel N-donor multimetallic $\text{M}-\text{M}'-\text{MOFs}$ [$\text{M}-\text{M}'(1,4-\text{bdc})_2(\text{dabco})\cdot4\text{DMF}\cdot1/2\text{H}_2\text{O}$, $\text{M}/\text{M}' = \text{Co, Ni, Cu}$. The $\text{M}-\text{M}'-\text{MOFs}$ were prepared by a facile mixed-metal approach. Typically, $\text{M}(\text{NO}_3)_2\cdot6\text{H}_2\text{O}$ and $\text{M}'(\text{NO}_3)_2\cdot6\text{H}_2\text{O}$, DABCO (triethylenediamine) and DMF were added to a sealed round-bottom flask. The mixture was treated by ultrasonic dispersion until the solids were dissolved, and then heated in an oil bath for 40 hours at 120 °C after stirring for 8 h. The products were washed with DMF and methanol, and then dried under vacuum at 150 °C for 12 h. The thermolysis was carried out in He with a heating rate of 1 °C min⁻¹. The samples were first heated from room temperature to 200 °C and then to 500 °C and kept at this temperature for 8 h to obtain $\text{M}-\text{M}'@C-\text{N}$ black catalysts. Because of the easy oxidation of metal NPs in air, the samples were reduced at 200 °C under a H_2 atmosphere for 2 h prior to being used as catalysts. As a result, the MOF-derived transition-metal alloy NPs were highly dispersed and embedded in the highly ordered N-doped graphene layers with an average size of *ca.* 20 nm. Moreover, the complete alloying of transition metal elements and the homogeneous distribution of N-doped carbon and metal alloy NPs was confirmed on every nanoparticle as well as and the existence of an obvious synergistic activation of different transition metals and strong coordination interactions between metals of alloy NPs and N atoms. When used in the transfer hydrogenation of nitriles in the absence of basic additives, $\text{Co-Ni}(3:1)@C-\text{N}$ showed the best catalytic performance being almost 5 times more active than its monometallic counterparts and showed excellent recyclability and extensive applicability.

The general approach in bimetallic MOFMS consists of (i) using Zn as the sacrificial secondary metal for its evaporation at high temperatures under inert conditions, thus improving the porosity of the resulting carbon based material and (ii) preparing bimetallic active sites/promoters as both unsupported and supported in carbon catalytic materials in oxidizing and inert atmospheres, respectively. Furthermore, bimetallic MOFMS allows more than that. During the thermal process that MOFMS basically is, the properties and behaviour of a certain metal will definitely affect the other metal's final state and *vice versa*, thus providing an extra control parameter to play with during this promising and relatively novel synthetic process.

4. Conclusions and future perspectives

Metal organic frameworks have emerged as exemplary precursors for the manufacture of highly catalytically active materials through the so-called MOF mediated synthesis. From

basic to advanced catalytic design, MOFMS embraces a wide range of synthetic techniques. Keeping it simple, the basic definition of MOFMS refers to a single-step heat treatment of the parent MOF. In this process, the properties of the MOF together with the heat treatment conditions (activating atmosphere, temperature, time and heating ramp) dictate the final properties of the MOF derived catalyst, allowing for an unprecedented freedom in design. As highlighted by the examples above, when obtained through MOFMS, metal oxides, carbons, carbides, and nanoparticles in carbon have unique structural properties that result, in many cases, in improved catalytic performance.

In an inert atmosphere, when the decomposition temperature of the framework is exceeded, the Gibbs free energy/reduction potential of metal ions predict the metallic species (metal, oxide, and carbide) which are likely to be formed. Once formed, the Tamman temperature of the metallic species needs to be considered in order to control the size of the resulting nanoparticles in the carbon matrix. Finally, according to the boiling point of the metal species, the metal phase will be retained in the material or, in contrast, will evaporate. Although we decided to organize this review according to the MOF metal component, the linker also plays a decisive role, not only defining the final porosity of the carbon matrix and its composition (oxidic, nitridic), but also the oxidation state of the metal species in the catalyst. For instance, oxygenated linkers will favour oxidation of species whereas nitrogen in the linker may result in the formation of highly dispersed M-N species in the final catalyst.

Calcination in air of the pristine MOF is a more straightforward procedure in the sense that it always leads to the formation of the corresponding carbon-free metal oxide. In this case, when the decomposition temperature of the framework is surpassed, the organic linker has been already burnt away. Further increasing the temperature allows the porosity of the material to be tuned. In some cases, a pyrolysis step prior to calcination is applied to enhance the porosity of the resulting material. The boiling point of oxides is usually too high to consider possible evaporation during MOFMS under an oxidizing atmosphere.

In conventional supported catalysts the most stable oxides such as MgO , ZrO_2 , Al_2O_3 , TiO_2 , SiO_2 , MnO , Cr_2O_3 , and ZnO act as supports and/or promoters. In MOFMS, these are achieved by calcination in air and also by pyrolysis in an inert atmosphere up to a given temperature, yielding non-supported and carbon supported metal oxides, respectively. In the latter case, according to their reduction potential and when oxygen available, these ions tend to form their corresponding oxides up to the temperature when carbon starts reducing the metal oxide to form the corresponding metal. However, this usually takes place at a relatively high temperature. These Mg, Zr, Al, Ti or Zn MOF derived catalysts are typically used as metal oxides for photocatalysis and/or as templates for preparing nanostructured carbons. In contrast, metals such as Fe, Co, Ni and Cu are likely to form metallic nanoparticles under an inert atmosphere at a relatively low temperature and these metals are highly effective in catalyzing the graphitization of carbon surrounding them,²⁶¹ which needs to be taken into consideration for the targeted catalytic application.



The biggest advantage of this new methodology lies in the number of degrees of freedom that it offers. When compared to traditional synthesis methods, the large metal dispersion in the original MOF allows for very high metal dispersion and for exquisite control over the final size of the supported/encapsulated nanoparticles. Moreover, further development and modification of the technique *via* utilization of different activation atmospheres, introduction of additional components to the MOF by ion exchange or impregnation, intermediate pyrolysis steps and others, has already led to sophisticated catalytic systems that cannot be achieved by any other means. This approach does not only lead to improved catalytic systems but is an excellent tool for the design of model systems that help generate knowledge through understanding.

Probably the aspect that seems to raise more concern regarding the MOFMS approach is the fact that highly engineered, beautiful, very often expensive, materials are simply destroyed and this may look like something unaffordable from an industrial point of view. However, this is not always the case. Some of the aforementioned MOF mediated materials are highly competitive from an economic point of view compared to other more sophisticated synthetic procedures and their counterparts obtained by conventional methods.⁹⁵ Moreover, keeping in mind a certain level of pragmatism, the main target of catalyst synthesis is the design and preparation of outstanding catalysts and MOFMS is an excellent tool to this end. On the other hand, one should also consider that a large part of the final price of a MOF comes from extensive washing steps needed to purify the as synthesized materials. Obviously, the nature and depth of these steps required for MOFMS are far milder and this should lead to moderate prices of the hard template. Moreover, if the stability of the resulting catalysts is increased by orders of magnitude, the impact in terms of catalyst makeup cost and less process operation interruptions has the potential of bringing additional economic attractiveness. Nevertheless, specific viability studies apply depending on the MOF precursor and synthesis conditions utilized in different cases.

Overall, we are certain that over the next few years this approach will be adopted by more research groups and that new methodologies will be developed. At this stage we safely state that our imagination is the only limit when it comes to the design possibilities in MOF mediated synthesis.

Acknowledgements

This project has received funding from the European Union's Horizon 2020 research and innovation programme under the Marie Skłodowska-Curie Grant Agreement No. 704473.

References

1. R. Jin, C. Zeng, M. Zhou and Y. Chen, Atomically Precise Colloidal Metal Nanoclusters and Nanoparticles: Fundamentals and Opportunities, *Chem. Rev.*, 2016, **116**(18), 10346–10413.
2. F. Akhtar, L. Andersson, S. Ogunwumi, N. Hedin and L. Bergström, Structuring adsorbents and catalysts by processing of porous powders, *J. Eur. Ceram. Soc.*, 2014, **34**(7), 1643–1666.
3. R. W. J. Scott, O. M. Wilson and R. M. Crooks, Synthesis, Characterization, and Applications of Dendrimer-Encapsulated Nanoparticles, *J. Phys. Chem. B*, 2005, **109**(2), 692–704.
4. A. E. Danks, S. R. Hall and Z. Schnepf, The evolution of 'sol-gel' chemistry as a technique for materials synthesis, *Mater. Horiz.*, 2016, **3**(2), 91–112.
5. R. W. Johnson, A. Hultqvist and S. F. Bent, A brief review of atomic layer deposition: from fundamentals to applications, *Mater. Today*, 2014, **17**(5), 236–246.
6. S. G. Kwon, G. Krylova, P. J. Phillips, R. F. Klie, S. Chatopadhyay and T. Shibata, *et al.*, Heterogeneous nucleation and shape transformation of multicomponent metallic nanostructures, *Nat. Mater.*, 2015, **14**(2), 215–223.
7. F. Zaera, Nanostructured materials for applications in heterogeneous catalysis, *Chem. Soc. Rev.*, 2013, **42**(7), 2746–2762.
8. J. Gascon, A. Corma, F. Kapteijn and F. X. Llabrés i Xamena, Metal Organic Framework Catalysis: Quo vadis?, *ACS Catal.*, 2014, **4**(2), 361–378.
9. H. Wang, Q.-L. Zhu, R. Zou and Q. Xu, Metal–Organic Frameworks for Energy Applications, *Chemistry*, 2017, **2**(1), 52–80.
10. M. G. Goesten, F. Kapteijn and J. Gascon, Fascinating chemistry or frustrating unpredictability: observations in crystal engineering of metal–organic frameworks, *CrystEngComm*, 2013, **15**(45), 9249–9257.
11. G. Férey, C. Mellot-Draznieks, C. Serre, F. Millange, J. Dutour, S. Surblé and I. Margioliaki, A chromium terephthalate-based solid with unusually large pore volumes and surface area, *Science*, 2005, **309**(5743), 2040–2042.
12. J. H. Cavka, S. Jakobsen, U. Olsbye, N. Guillou, C. Lamberti and S. Bordiga, *et al.*, A New Zirconium Inorganic Building Brick Forming Metal Organic Frameworks with Exceptional Stability, *J. Am. Chem. Soc.*, 2008, **130**(42), 13850–13851.
13. V. Colombo, S. Galli, H. J. Choi, G. D. Han, A. Maspero and G. Palmisano, *et al.*, High thermal and chemical stability in pyrazolate-bridged metal–organic frameworks with exposed metal sites, *Chem. Sci.*, 2011, **2**(7), 1311–1319.
14. Z.-W. Wang, M. Chen, C.-S. Liu, X. Wang, H. Zhao and M. Du, A Versatile Al^{III}-Based Metal–Organic Framework with High Physicochemical Stability, *Chem. – Eur. J.*, 2015, **21**(48), 17215–17219.
15. B. Liu, H. Shioyama, T. Akita and Q. Xu, Metal–Organic Framework as a Template for Porous Carbon Synthesis, *J. Am. Chem. Soc.*, 2008, **130**(16), 5390–5391.
16. L. Zhang, C. Feng, S. Gao, Z. Wang and C. Wang, Palladium nanoparticle supported on metal–organic framework derived N-decorated nanoporous carbon as an efficient catalyst for the Suzuki coupling reaction, *Catal. Commun.*, 2015, **61**, 21–25.
17. W. Zhong, H. Liu, C. Bai, S. Liao and Y. Li, Base-Free Oxidation of Alcohols to Esters at Room Temperature and Atmospheric Conditions using Nanoscale Co-Based Catalysts, *ACS Catal.*, 2015, **5**(3), 1850–1856.
18. K. Shen, L. Chen, J. Long, W. Zhong and Y. Li, MOFs-Templated Co@Pd Core–Shell NPs Embedded in N-Doped



Carbon Matrix with Superior Hydrogenation Activities, *ACS Catal.*, 2015, **5**(9), 5264–5271.

19 I. Mondal and U. Pal, Synthesis of MOF templated Cu/CuO@TiO₂ nanocomposites for synergistic hydrogen production, *Phys. Chem. Chem. Phys.*, 2016, **18**(6), 4780–4788.

20 R. Li, S. Wu, X. Wan, H. Xu and Y. Xiong, Cu/TiO₂ octahedral-shell photocatalysts derived from metal-organic framework@semiconductor hybrid structures, *Inorg. Chem. Front.*, 2016, **3**(1), 104–110.

21 Z. Dong, X. Le, Y. Liu, C. Dong and J. Ma, Metal organic framework derived magnetic porous carbon composite supported gold and palladium nanoparticles as highly efficient and recyclable catalysts for reduction of 4-nitrophenol and hydrodechlorination of 4-chlorophenol, *J. Mater. Chem. A*, 2014, **2**(44), 18775–18785.

22 V. P. Santos, T. A. Wezendonk, J. J. D. Jaén, A. I. Dugulan, M. A. Nasalevich and H.-U. Islam, *et al.*, Metal organic framework-mediated synthesis of highly active and stable Fischer-Tropsch catalysts, *Nat. Commun.*, 2015, **6**, 1–8.

23 D. Chen, M. Huang, S. He, S. He, L. Ding and Q. Wang, *et al.*, Ru-MOF enwrapped by montmorillonite for catalyzing benzene hydrogenation, *Appl. Clay Sci.*, 2016, **119**(Part 1), 109–115.

24 X. Cao, B. Zheng, X. Rui, W. Shi, Q. Yan and H. Zhang, Metal Oxide-Coated Three-Dimensional Graphene Prepared by the Use of Metal–Organic Frameworks as Precursors, *Angew. Chem., Int. Ed.*, 2014, **53**(5), 1404–1409.

25 C. D. Malonzo, S. M. Shaker, L. Ren, S. D. Prinslow, A. E. Platero-Prats and L. C. Gallington, *et al.*, Thermal Stabilization of Metal–Organic Framework-Derived Single-Site Catalytic Clusters Through Nanocasting, *J. Am. Chem. Soc.*, 2016, **138**(8), 2739–2748.

26 L. Zhang, X. Wang, R. Wang and M. Hong, Structural Evolution from Metal–Organic Framework to Hybrids of Nitrogen-Doped Porous Carbon and Carbon Nanotubes for Enhanced Oxygen Reduction Activity, *Chem. Mater.*, 2015, **27**(22), 7610–7618.

27 S. Deng, M. Kurttepeli, D. J. Cott, S. Bals and C. Detavernier, Porous nanostructured metal oxides synthesized through atomic layer deposition on a carbonaceous template followed by calcination, *J. Mater. Chem. A*, 2015, **3**(6), 2642–2649.

28 M. Enterría and J. L. Figueiredo, Nanostructured mesoporous carbons: Tuning texture and surface chemistry, *Carbon*, 2016, **108**, 79–102.

29 C. T. Campbell, S. C. Parker and D. E. Starr, The Effect of Size-Dependent Nanoparticle Energetics on Catalyst Sintering, *Science*, 2002, **298**(5594), 811–814.

30 J.-K. Sun and Q. Xu, Functional materials derived from open framework templates/precursors: synthesis and applications, *Energy Environ. Sci.*, 2014, **7**(7), 2071–2100.

31 W. Xia, A. Mahmood, R. Zou and Q. Xu, Metal–organic frameworks and their derived nanostructures for electrochemical energy storage and conversion, *Energy Environ. Sci.*, 2015, **8**(7), 1837–1866.

32 H. J. T. Ellingham, Reducibility of oxides and sulfides in metallurgical processes, *J. Soc. Chem. Ind., London*, 1944, **63**, 9.

33 R. Das, P. Pachfule, R. Banerjee and P. Poddar, Metal and metal oxide nanoparticle synthesis from metal organic frameworks (MOFs), finding the border of metal and metal oxides, *Nanoscale*, 2012, **4**(2), 591–599.

34 N. Birks, G. H. Meier and F. S. Pettit, *Introduction to the High Temperature Oxidation of Metals*, Cambridge University Press, Cambridge, 2nd edn, 2006, 2006/003/17.

35 Elements MaBPOC. http://www.chemistry.patent-invent.com/chemistry/melting_boiling_point.html, updated April 2011.

36 L. B. Railsback, <http://www.gly.uga.edu/railsback/Fundamentals/ElementalAbundanceTableP.pdf>, 2006.

37 M. Trueba and S. P. Trasatti, γ -Alumina as a Support for Catalysts: A Review of Fundamental Aspects, *Eur. J. Inorg. Chem.*, 2005, 3393–3403.

38 T. Ahnfeldt, N. Guillou, D. Gunzelmann, I. Margiolas, T. Loiseau and G. Férey, *et al.*, $[\text{Al}_4(\text{OH})_2(\text{OCH}_3)_4(\text{H}_2\text{N}-\text{bdc})_3] \cdot x\text{H}_2\text{O}$: A 12-Connected Porous Metal–Organic Framework with an Unprecedented Aluminum-Containing Brick, *Angew. Chem., Int. Ed.*, 2009, **48**(28), 5163–5166.

39 E. D. Bloch, D. Britt, C. Lee, C. J. Doonan, F. J. Uribe-Romo and H. Furukawa, *et al.*, Metal Insertion in a Microporous Metal–Organic Framework Lined with 2,2'-Bipyridine, *J. Am. Chem. Soc.*, 2010, **132**(41), 14382–14384.

40 F. Afsahi, H. Vinh-Thang, S. Mikhailenko and S. Kaliaguine, Electrocatalyst synthesized from metal organic frameworks, *J. Power Sources*, 2013, **239**, 415–423.

41 Q.-L. Zhu, W. Xia, T. Akita, R. Zou and Q. Xu, Metal–Organic Framework-Derived Honeycomb-Like Open Porous Nanostructures as Precious-Metal-Free Catalysts for Highly Efficient Oxygen Electroreduction, *Adv. Mater.*, 2016, **28**(30), 6391–6398.

42 S. S. Shendage, A. S. Singh and J. M. Nagarkar, Facile approach to the electrochemical synthesis of palladium-reduced graphene oxide and its application for Suzuki coupling reaction, *Tetrahedron Lett.*, 2014, **55**(4), 857–860.

43 H.-L. Jiang, B. Liu, Y.-Q. Lan, K. Kuratani, T. Akita and H. Shioyama, *et al.*, From Metal–Organic Framework to Nanoporous Carbon: Toward a Very High Surface Area and Hydrogen Uptake, *J. Am. Chem. Soc.*, 2011, **133**(31), 11854–11857.

44 M. Schubert, U. Müller and N. Trukhan, *Metal oxides from metal–organic framework materials*, Google Patents, US 8501150 B2, 2013.

45 M. Celebi, M. Yurderi, A. Bulut, M. Kaya and M. Zahmakiran, Palladium nanoparticles supported on amine-functionalized SiO₂ for the catalytic hexavalent chromium reduction, *Appl. Catal., B*, 2016, **180**, 53–64.

46 M. Yadav and Q. Xu, Catalytic chromium reduction using formic acid and metal nanoparticles immobilized in a metal–organic framework, *Chem. Commun.*, 2013, **49**(32), 3327–3329.

47 R. Cheng, Z. Liu, L. Zhong, X. He, P. Qiu and M. Terano, *et al.*, Phillips Cr/Silica Catalyst for Ethylene Polymerization, in *Polyolefins: 50 years after Ziegler and Natta I: Polyethylene and Polypropylene*, ed. W. Kaminsky, Berlin, Heidelberg, Springer Berlin Heidelberg, 2013, pp. 135–202.



48 A. Budnyk, A. Damin, C. Barzan, E. Groppo, C. Lamberti and S. Bordiga, *et al.*, Cr-doped porous silica glass as a model material to describe Phillips catalyst properties, *J. Catal.*, 2013, **308**, 319–327.

49 D.-W. Lee, M. S. Lee, J. Y. Lee, S. Kim, H.-J. Eom and D. J. Moon, *et al.*, The review of Cr-free Fe-based catalysts for high-temperature water-gas shift reactions, *Catal. Today*, 2013, **210**, 2–9.

50 H. Zhao, H. Song, L. Xu and L. Chou, Isobutane dehydrogenation over the mesoporous $\text{Cr}_2\text{O}_3/\text{Al}_2\text{O}_3$ catalysts synthesized from a metal-organic framework MIL-101, *Appl. Catal., A*, 2013, **456**, 188–196.

51 X. Qiu, X. Wang and Y. Li, Controlled growth of dense and ordered metal-organic framework nanoparticles on graphene oxide, *Chem. Commun.*, 2015, **51**(18), 3874–3877.

52 S. Bhattacharjee, C. Chen and W.-S. Ahn, Chromium terephthalate metal-organic framework MIL-101: synthesis, functionalization, and applications for adsorption and catalysis, *RSC Adv.*, 2014, **4**(94), 52500–52525.

53 G. Wang, C. Li and H. Shan, Highly Efficient Metal Sulfide Catalysts for Selective Dehydrogenation of Isobutane to Isobutene, *ACS Catal.*, 2014, **4**(4), 1139–1143.

54 C.-B. Wang, C.-C. Lee, J.-L. Bi, J.-Y. Siang, J.-Y. Liu and C.-T. Yeh, Study on the steam reforming of ethanol over cobalt oxides, *Catal. Today*, 2009, **146**(1–2), 76–81.

55 A. Y. Khodakov, W. Chu and P. Fongarland, Advances in the Development of Novel Cobalt Fischer-Tropsch Catalysts for Synthesis of Long-Chain Hydrocarbons and Clean Fuels, *Chem. Rev.*, 2007, **107**(5), 1692–1744.

56 F. Hebrard and P. Kalck, Cobalt-Catalyzed Hydroformylation of Alkenes: Generation and Recycling of the Carbonyl Species, and Catalytic Cycle, *Chem. Rev.*, 2009, **109**(9), 4272–4282.

57 R. Banerjee, A. Phan, B. Wang, C. Knobler, H. Furukawa and M. O'Keeffe, *et al.*, High-Throughput Synthesis of Zeolitic Imidazolate Frameworks and Application to CO_2 Capture, *Science*, 2008, **319**(5865), 939–943.

58 J. Qian, F. Sun and L. Qin, Hydrothermal synthesis of zeolitic imidazolate framework-67 (ZIF-67) nanocrystals, *Mater. Lett.*, 2012, **82**, 220–223.

59 H. Jing, X. Song, S. Ren, Y. Shi, Y. An and Y. Yang, *et al.*, ZIF-67 Derived Nanostructures of Co/CoO and Co@N-doped Graphitic Carbon as Counter Electrode for Highly Efficient Dye-sensitized Solar Cells, *Electrochim. Acta*, 2016, **213**, 252–259.

60 Y.-Q. Tian, Z.-X. Chen, L.-H. Weng, H.-B. Guo, S. Gao and D. Y. Zhao, Two Polymorphs of Cobalt(II) Imidazolate Polymers Synthesized Solvothermally by Using One Organic Template *N,N*-Dimethylacetamide, *Inorg. Chem.*, 2004, **43**(15), 4631–4635.

61 S. Ma, G. A. Goenaga, A. V. Call and D.-J. Liu, Cobalt Imidazolate Framework as Precursor for Oxygen Reduction Reaction Electrocatalysts, *Chem. – Eur. J.*, 2011, **17**(7), 2063–2067.

62 F. Jaouen, J. Herranz, M. Lefèvre, J.-P. Dodelet, U. I. Kramm and I. Herrmann, *et al.*, Cross-Laboratory Experimental Study of Non-Noble-Metal Electrocatalysts for the Oxygen Reduction Reaction, *ACS Appl. Mater. Interfaces*, 2009, **1**(8), 1623–1639.

63 J. R. Pels, F. Kapteijn, J. A. Moulijn, Q. Zhu and K. M. Thomas, Evolution of nitrogen functionalities in carbonaceous materials during pyrolysis, *Carbon*, 1995, **33**(11), 1641–1653.

64 W. Xia, J. Zhu, W. Guo, L. An, D. Xia and R. Zou, Well-defined carbon polyhedrons prepared from nano metal-organic frameworks for oxygen reduction, *J. Mater. Chem. A*, 2014, **2**(30), 11606–11613.

65 W. Xia, R. Zou, L. An, D. Xia and S. Guo, A metal-organic framework route to *in situ* encapsulation of $\text{Co}@\text{Co}_3\text{O}_4@\text{C}$ core@bisphere nanoparticles into a highly ordered porous carbon matrix for oxygen reduction, *Energy Environ. Sci.*, 2015, **8**(2), 568–576.

66 X. Wang, J. Zhou, H. Fu, W. Li, X. Fan and G. Xin, *et al.*, MOF derived catalysts for electrochemical oxygen reduction, *J. Mater. Chem. A*, 2014, **2**(34), 14064–14070.

67 A. Kong, C. Mao, Q. Lin, X. Wei, X. Bu and P. Feng, From cage-in-cage MOF to N-doped and Co-nanoparticle-embedded carbon for oxygen reduction reaction, *Dalton Trans.*, 2015, **44**(15), 6748–6754.

68 S.-T. Zheng, T. Wu, B. Irfanoglu, F. Zuo, P. Feng and X. Bu, Multicomponent Self-Assembly of a Nested $\text{Co}_{24}@\text{Co}_{48}$ Metal-Organic Polyhedral Framework, *Angew. Chem., Int. Ed.*, 2011, **50**(35), 8034–8037.

69 S. H. Oh, R. Black, E. Pomerantseva, J.-H. Lee and L. F. Nazar, Synthesis of a metallic mesoporous pyrochlore as a catalyst for lithium– O_2 batteries, *Nat. Chem.*, 2012, **4**(12), 1004–1010.

70 T. Y. Ma, S. Dai, M. Jaroniec and S. Z. Qiao, Metal-Organic Framework Derived Hybrid Co_3O_4 -Carbon Porous Nanowire Arrays as Reversible Oxygen Evolution Electrodes, *J. Am. Chem. Soc.*, 2014, **136**(39), 13925–13931.

71 S. Bao, N. Yan, X. Shi, R. Li and Q. Chen, High and stable catalytic activity of porous $\text{Ag}/\text{Co}_3\text{O}_4$ nanocomposites derived from MOFs for CO oxidation, *Appl. Catal., A*, 2014, **487**, 189–194.

72 X. Wang, W. Zhong and Y. Li, Nanoscale Co-based catalysts for low-temperature CO oxidation, *Catal. Sci. Technol.*, 2015, **5**(2), 1014–1020.

73 X. Xie, Y. Li, Z.-Q. Liu, M. Haruta and W. Shen, Low-temperature oxidation of CO catalysed by Co_3O_4 nanorods, *Nature*, 2009, **458**(7239), 746–749.

74 I. Stará, V. Nehasil and V. Matolín, The influence of particle size on CO oxidation on Pd/alumina model catalyst, *Surf. Sci.*, 1995, **331**–333(Part A), 173–177.

75 M. Haruta, S. Tsubota, T. Kobayashi, H. Kageyama, M. J. Genet and B. Delmon, Low-Temperature Oxidation of CO over Gold Supported on TiO_2 , $\alpha\text{-Fe}_2\text{O}_3$, and Co_3O_4 , *J. Catal.*, 1993, **144**(1), 175–192.

76 H. Einaga, S. Futamura and T. Ibusuki, Complete Oxidation of Benzene in Gas Phase by Platinized Titania Photocatalysts, *Environ. Sci. Technol.*, 2001, **35**(9), 1880–1884.

77 J. Li, W. Li, G. Liu, Y. Deng, J. Yang and Y. Chen, Tricobalt tetraoxide-supported palladium catalyst derived from



metal organic frameworks for complete benzene oxidation, *Catal. Lett.*, 2016, **146**(7), 1300–1308.

78 Y.-X. Zhou, Y.-Z. Chen, L. Cao, J. Lu and H.-L. Jiang, Conversion of a metal-organic framework to N-doped porous carbon incorporating Co and CoO nanoparticles: direct oxidation of alcohols to esters, *Chem. Commun.*, 2015, **51**(39), 8292–8295.

79 C. Bai, X. Yao and Y. Li, Easy Access to Amides through Aldehydic C–H Bond Functionalization Catalyzed by Heterogeneous Co-Based Catalysts, *ACS Catal.*, 2015, **5**(2), 884–891.

80 K.-Y. Andrew Lin, F.-K. Hsu and W.-D. Lee, Magnetic cobalt-graphene nanocomposite derived from self-assembly of MOFs with graphene oxide as an activator for peroxymonosulfate, *J. Mater. Chem. A*, 2015, **3**(18), 9480–9490.

81 C. Xing, Y. Liu, Y. Su, Y. Chen, S. Hao and X. Wu, *et al.*, Structural Evolution of Co-Based Metal Organic Frameworks in Pyrolysis for Synthesis of Core–Shells on Nanosheets: Co@Co_x@Carbon-rGO Composites for Enhanced Hydrogen Generation Activity, *ACS Appl. Mater. Interfaces*, 2016, **8**(24), 15430–15438.

82 S. Enthaler, D. Addis, K. Junge, G. Erre and M. Beller, A General and Environmentally Benign Catalytic Reduction of Nitriles to Primary Amines, *Chem. – Eur. J.*, 2008, **14**(31), 9491–9494.

83 D. Pingen, C. Müller and D. Vogt, Direct Amination of Secondary Alcohols Using Ammonia, *Angew. Chem., Int. Ed.*, 2010, **49**(44), 8130–8133.

84 J. Long, Y. Zhou and Y. Li, Transfer hydrogenation of unsaturated bonds in the absence of base additives catalyzed by a cobalt-based heterogeneous catalyst, *Chem. Commun.*, 2015, **51**(12), 2331–2334.

85 Z.-R. Dong, Y.-Y. Li, J.-S. Chen, B.-Z. Li, Y. Xing and J.-X. Gao, Highly Efficient Iridium Catalyst for Asymmetric Transfer Hydrogenation of Aromatic Ketones under Base-Free Conditions, *Org. Lett.*, 2005, **7**(6), 1043–1045.

86 X. Wang and Y. Li, Chemoselective hydrogenation of functionalized nitroarenes using MOF-derived co-based catalysts, *J. Mol. Catal. A: Chem.*, 2016, **420**, 56–65.

87 X. Ma, Y.-X. Zhou, H. Liu, Y. Li and H.-L. Jiang, A MOF-derived Co–CoO@N-doped porous carbon for efficient tandem catalysis: dehydrogenation of ammonia borane and hydrogenation of nitro compounds, *Chem. Commun.*, 2016, **52**(49), 7719–7722.

88 B. Chen, Z. Yang, Y. Zhu and Y. Xia, Zeolitic imidazolate framework materials: recent progress in synthesis and applications, *J. Mater. Chem. A*, 2014, **2**(40), 16811–16831.

89 M. B. Gawande, A. Goswami, F.-X. Felpin, T. Asefa, X. Huang and R. Silva, *et al.*, Cu and Cu-Based Nanoparticles: Synthesis and Applications in Catalysis, *Chem. Rev.*, 2016, **116**(6), 3722–3811.

90 S. S.-Y. Chui, S. M.-F. Lo, J. P. H. Charmant, A. G. Orpen and I. D. Williams, A Chemically Functionalizable Nanoporous Material [Cu₃(TMA)₂(H₂O)₃]_n, *Science*, 1999, **283**(5405), 1148–1150.

91 K.-S. Lin, A. K. Adhikari, C.-N. Ku, C.-L. Chiang and H. Kuo, Synthesis and characterization of porous HKUST-1 metal organic frameworks for hydrogen storage, *Int. J. Hydrogen Energy*, 2012, **37**(18), 13865–13871.

92 C. G. Carson, K. Hardcastle, J. Schwartz, X. Liu, C. Hoffmann and R. A. Gerhardt, *et al.*, Synthesis and Structure Characterization of Copper Terephthalate Metal–Organic Frameworks, *Eur. J. Inorg. Chem.*, 2009, 2338–2343.

93 H. K. Kim, W. S. Yun, M.-B. Kim, J. Y. Kim, Y.-S. Bae and J. Lee, *et al.*, A Chemical Route to Activation of Open Metal Sites in the Copper-Based Metal–Organic Framework Materials HKUST-1 and Cu-MOF-2, *J. Am. Chem. Soc.*, 2015, **137**(31), 10009–10015.

94 C. Xu, Y. Zhang, L. Wang, L. Xu, X. Bian and H. Ma, *et al.*, Nanotubular Mesoporous PdCu Bimetallic Electrocatalysts toward Oxygen Reduction Reaction, *Chem. Mater.*, 2009, **21**(14), 3110–3116.

95 Y. Luo, L. A. Estudillo-Wong, L. Cavillo, G. Granozzi and N. Alonso-Vante, An easy and cheap chemical route using a MOF precursor to prepare Pd–Cu electrocatalyst for efficient energy conversion cathodes, *J. Catal.*, 2016, **338**, 135–142.

96 J.-B. Raoof, S. R. Hosseini, R. Ojani and S. Mandegarzad, MOF-derived Cu/nanoporous carbon composite and its application for electro-catalysis of hydrogen evolution reaction, *Energy*, 2015, **90**(Part 1), 1075–1081.

97 Z. Xiang, D. Cao, X. Shao, W. Wang, J. Zhang and W. Wu, Facile preparation of high-capacity hydrogen storage metal–organic frameworks: A combination of microwave-assisted solvothermal synthesis and supercritical activation, *Chem. Eng. Sci.*, 2010, **65**(10), 3140–3146.

98 A. Razeghi, A. Khodadadi, H. Ziae-Azad and Y. Mortazavi, Activity enhancement of Cu-doped ceria by reductive regeneration of CuO–CeO₂ catalyst for preferential oxidation of CO in H₂-rich streams, *Chem. Eng. J.*, 2010, **164**(1), 214–220.

99 J. M. Zamaro, N. C. Pérez, E. E. Miró, C. Casado, B. Seoane and C. Téllez, *et al.*, HKUST-1 MOF: A matrix to synthesize CuO and CuO–CeO₂ nanoparticle catalysts for CO oxidation, *Chem. Eng. J.*, 2012, **195–196**, 180–187.

100 S. Qian, C. Wang, W. Liu, Y. Zhu, W. Yao and X. Lu, An enhanced Cds/TiO₂ photocatalyst with high stability and activity: Effect of mesoporous substrate and bifunctional linking molecule, *J. Mater. Chem.*, 2011, **21**(13), 4945–4952.

101 H. Liu, S. Zhang, Y. Liu, Z. Yang, X. Feng and X. Lu, *et al.*, Well-Dispersed and Size-Controlled Supported Metal Oxide Nanoparticles Derived from MOF Composites and Further Application in Catalysis, *Small*, 2015, **11**(26), 3130–3134.

102 H. Niu, S. Liu, Y. Cai, F. Wu and X. Zhao, MOF derived porous carbon supported Cu/Cu₂O composite as high performance non-noble catalyst, *Microporous Mesoporous Mater.*, 2016, **219**, 48–53.

103 T. Zeng, X. Zhang, S. Wang, Y. Ma, H. Niu and Y. Cai, A double-shelled yolk-like structure as an ideal magnetic support of tiny gold nanoparticles for nitrophenol reduction, *J. Mater. Chem. A*, 2013, **1**(38), 11641–11647.

104 T. Wu, L. Zhang, J. Gao, Y. Liu, C. Gao and J. Yan, Fabrication of graphene oxide decorated with Au–Ag alloy



nanoparticles and its superior catalytic performance for the reduction of 4-nitrophenol, *J. Mater. Chem. A*, 2013, **1**(25), 7384–7390.

105 S. Gao, X. Jia, Z. Li and Y. Chen, Hierarchical plasmonic-metal/semiconductor micro/nanostructures: green synthesis and application in catalytic reduction of *p*-nitrophenol, *J. Nanopart. Res.*, 2012, **14**(3), 1–11.

106 B. R. Kim, J. S. Oh, J. Kim and C. Y. Lee, Robust Aerobic Alcohol Oxidation Catalyst Derived from Metal–Organic Frameworks, *Catal. Lett.*, 2016, **146**(4), 734–743.

107 Y. Yang, Z. Du, Y. Huang, F. Lu, F. Wang and J. Gao, *et al.*, Conversion of furfural into cyclopentanone over Ni–Cu bimetallic catalysts, *Green Chem.*, 2013, **15**(7), 1932–1940.

108 Y. Wang, S. Sang, W. Zhu, L. Gao and G. Xiao, CuNi@C catalysts with high activity derived from metal–organic frameworks precursor for conversion of furfural to cyclopentanone, *Chem. Eng. J.*, 2016, **299**, 104–111.

109 M. Munoz, P. Domínguez, Z. M. de Pedro, J. A. Casas and J. J. Rodriguez, Naturally-occurring iron minerals as inexpensive catalysts for CWPO, *Appl. Catal., B*, 2017, **203**, 166–173.

110 K. E. deKrafft, C. Wang and W. Lin, Metal–Organic Framework Templated Synthesis of $\text{Fe}_2\text{O}_3/\text{TiO}_2$ Nanocomposite for Hydrogen Production, *Adv. Mater.*, 2012, **24**(15), 2014–2018.

111 S.-w. Lee, J. Drwiega, C.-Y. Wu, D. Mazyck and W. M. Sigmund, Anatase TiO_2 Nanoparticle Coating on Barium Ferrite Using Titanium Bis-Ammonium Lactato Dihydroxide and Its Use as a Magnetic Photocatalyst, *Chem. Mater.*, 2004, **16**(6), 1160–1164.

112 E. Kim and M. Yoon, Facile synthesis of $\gamma\text{-Fe}_2\text{O}_3$ @porous carbon materials using an Fe-based metal–organic framework: structure and porosity study, *J. Porous Mater.*, 2015, **22**(6), 1495–1502.

113 A. Banerjee, R. Gokhale, S. Bhatnagar, J. Jog, M. Bhardwaj and B. Lefez, *et al.*, MOF derived porous carbon- Fe_3O_4 nanocomposite as a high performance, recyclable environmental super-adsorbent, *J. Mater. Chem.*, 2012, **22**(37), 19694–19699.

114 F.-X. Qin, S.-Y. Jia, Y. Liu, X. Han, H.-T. Ren and W.-W. Zhang, *et al.*, Metal–organic framework as a template for synthesis of magnetic CoFe_2O_4 nanocomposites for phenol degradation, *Mater. Lett.*, 2013, **101**, 93–95.

115 J. Li, C.-Y. Liu and Y. Liu, Au/graphene hydrogel: synthesis, characterization and its use for catalytic reduction of 4-nitrophenol, *J. Mater. Chem.*, 2012, **22**(17), 8426–8430.

116 Y. Li, Y.-X. Zhou, X. Ma and H.-L. Jiang, A metal–organic framework-templated synthesis of [gamma]- Fe_2O_3 nanoparticles encapsulated in porous carbon for efficient and chemoselective hydrogenation of nitro compounds, *Chem. Commun.*, 2016, **52**, 4199–4202.

117 H. J. Lee, W. Cho, E. Lim and M. Oh, One-pot synthesis of magnetic particle-embedded porous carbon composites from metal–organic frameworks and their sorption properties, *Chem. Commun.*, 2014, **50**(41), 5476–5479.

118 A. Morozan, M. T. Sougrati, V. Goellner, D. Jones, L. Stievano and F. Jaouen, Effect of Furfuryl Alcohol on Metal Organic Framework-based Fe/N/C Electrocatalysts for Polymer Electrolyte Membrane Fuel Cells, *Electrochim. Acta*, 2014, **119**, 192–205.

119 G. Zhang, R. Chenitz, M. Lefèvre, S. Sun and J.-P. Dodelet, Is iron involved in the lack of stability of Fe/N/C electrocatalysts used to reduce oxygen at the cathode of PEM fuel cells?, *Nano Energy*, 2016, **29**, 111–125.

120 L. Yang, N. Larouche, R. Chenitz, G. Zhang, M. Lefèvre and J.-P. Dodelet, Activity, Performance, and Durability for the Reduction of Oxygen in PEM Fuel Cells, of Fe/N/C Electrocatalysts Obtained from the Pyrolysis of Metal–Organic-Framework and Iron Porphyrin Precursors, *Electrochim. Acta*, 2015, **159**, 184–197.

121 C. Mao, A. Kong, Y. Wang, X. Bu and P. Feng, MIL-100 derived nitrogen-embodied carbon shells embedded with iron nanoparticles, *Nanoscale*, 2015, **7**(24), 10817–10822.

122 F. Afsahi and S. Kaliaguine, Non-precious electrocatalysts synthesized from metal–organic frameworks, *J. Mater. Chem. A*, 2014, **2**(31), 12270–12279.

123 E. Proietti, F. Jaouen, M. Lefèvre, N. Larouche, J. Tian and J. Herranz, *et al.*, Iron-based cathode catalyst with enhanced power density in polymer electrolyte membrane fuel cells, *Nat. Commun.*, 2011, **2**, 416.

124 J.-S. Li, S.-L. Li, Y.-J. Tang, M. Han, Z.-H. Dai and J.-C. Bao, *et al.*, Nitrogen-doped $\text{Fe}/\text{Fe}_3\text{C}$ @graphitic layer/carbon nanotube hybrids derived from MOFs: efficient bifunctional electrocatalysts for ORR and OER, *Chem. Commun.*, 2015, **51**(13), 2710–2713.

125 S. Zhao, H. Yin, L. Du, L. He, K. Zhao and L. Chang, *et al.*, Carbonized Nanoscale Metal–Organic Frameworks as High Performance Electrocatalyst for Oxygen Reduction Reaction, *ACS Nano*, 2014, **8**(12), 12660–12668.

126 V. P. Santos, T. A. Wezendonk, J. J. D. Jaén, A. I. Dugulan, M. A. Nasalevich and H.-U. Islam, *et al.*, Metal organic framework-mediated synthesis of highly active and stable Fischer-Tropsch catalysts, *Nat. Commun.*, 2015, **6**, 1–8.

127 T. A. Wezendonk, V. P. Santos, M. A. Nasalevich, Q. S. E. Warringa, A. I. Dugulan and A. Chojecki, *et al.*, Elucidating the Nature of Fe Species during Pyrolysis of the Fe-BTC MOF into Highly Active and Stable Fischer-Tropsch Catalysts, *ACS Catal.*, 2016, 3236–3247.

128 B. An, K. Cheng, C. Wang, Y. Wang and W. Lin, Pyrolysis of Metal–Organic Frameworks to $\text{Fe}_3\text{O}_4@\text{Fe}_5\text{C}_2$ Core–Shell Nanoparticles for Fischer-Tropsch Synthesis, *ACS Catal.*, 2016, **6**(6), 3610–3618.

129 T. Wezendonk, Q. S. E. Warringa, V. Da Costa Oliveira Santos Silva, A. Chojecki, M. Ruitenbeek and G. Meima, *et al.*, Structural and elemental influence from various MOFs on the performance of Fe@C catalysts for Fischer-Tropsch synthesis, *Faraday Discuss.*, 2017, DOI: 10.1039/C6FD00198J.

130 W. Peng, J. Li, B. Chen, N. Wang, G. Luo and F. Wei, Mesoporous MgO synthesized by a homogeneous-hydrothermal method and its catalytic performance on gas-phase acetone condensation at low temperatures, *Catal. Commun.*, 2016, **74**, 39–42.



131 A. Mallick, S. Saha, P. Pachfule, S. Roy and R. Banerjee, Selective CO₂ and H₂ adsorption in a chiral magnesium-based metal organic framework (Mg-MOF) with open metal sites, *J. Mater. Chem.*, 2010, **20**(41), 9073–9080.

132 N. Salehifar, Z. Zarghami and M. Ramezani, A facile, novel and low-temperature synthesis of MgO nanorods *via* thermal decomposition using new starting reagent and its photocatalytic activity evaluation, *Mater. Lett.*, 2016, **167**, 226–229.

133 K. Mageshwari, S. S. Mali, R. Sathyamoorthy and P. S. Patil, Template-free synthesis of MgO nanoparticles for effective photocatalytic applications, *Powder Technol.*, 2013, **249**, 456–462.

134 A. Feinle, A. Heugenhauser and N. Hüsing, Impact of surfactants and acids on the sol-gel synthesis of MgO aerogels, *J. Supercrit. Fluids*, 2015, **106**, 133–139.

135 A. Hazra Chowdhury, I. Hazra Chowdhury and M. Kanti Naskar, A facile synthesis of grainy rod-like porous MgO, *Mater. Lett.*, 2015, **158**, 190–193.

136 A. J. Brown, P. S. Francis, J. L. Adcock, K. F. Lim and N. W. Barnett, Manganese(III) and manganese(IV) as chemiluminescence reagents: A review, *Anal. Chim. Acta*, 2008, **624**(2), 175–183.

137 N. D. Wasalathanthri, T. M. SantaMaria, D. A. Kriz, S. L. Dissanayake, C.-H. Kuo and S. Biswas, *et al.*, Mesoporous manganese oxides for NO₂ assisted catalytic soot oxidation, *Appl. Catal., B*, 2017, **201**, 543–551.

138 H. Einaga, S. Yamamoto, N. Maeda and Y. Teraoka, Structural analysis of manganese oxides supported on SiO₂ for benzene oxidation with ozone, *Catal. Today*, 2015, **242**(Part B), 287–293.

139 L. Peng, J. Zhang, Z. Xue, B. Han, J. Li and G. Yang, Large-pore mesoporous Mn₃O₄ crystals derived from metal-organic frameworks, *Chem. Commun.*, 2013, **49**(99), 11695–11697.

140 P. Zhang, Y. Zhan, B. Cai, C. Hao, J. Wang and C. Liu, *et al.*, Shape-controlled synthesis of Mn₃O₄ nanocrystals and their catalysis of the degradation of methylene blue, *Nano Res.*, 2010, **3**(4), 235–243.

141 N. M. Al-Mansi and N. M. Abdel Monem, Recovery of nickel oxide from spent catalyst, *Waste Manage.*, 2002, **22**(1), 85–90.

142 S. Z. Tasker, E. A. Standley and T. F. Jamison, Recent advances in homogeneous nickel catalysis, *Nature*, 2014, **509**(7500), 299–309.

143 Z. Zhang, L. Han, R. Chai, Q. Zhang, Y. Li and G. Zhao, *et al.*, Microstructured CeO₂-NiO-Al₂O₃/Ni-foam catalyst for oxidative dehydrogenation of ethane to ethylene, *Catal. Commun.*, 2017, **88**, 90–93.

144 R. K. Singha, A. Shukla, A. Yadav, L. N. Sivakumar Konathala and R. Bal, Effect of metal-support interaction on activity and stability of Ni-CeO₂ catalyst for partial oxidation of methane, *Appl. Catal., B*, 2017, **202**, 473–488.

145 X. Li, D. Li, H. Tian, L. Zeng, Z.-J. Zhao and J. Gong, Dry reforming of methane over Ni/La₂O₃ nanorod catalysts with stabilized Ni nanoparticles, *Appl. Catal., B*, 2017, **202**, 683–694.

146 S. Rakass, H. Oudghiri-Hassani, P. Rowntree and N. Abatzoglou, Steam reforming of methane over unsupported nickel catalysts, *J. Power Sources*, 2006, **158**(1), 485–496.

147 J. Sehested, S. Dahl, J. Jacobsen and J. R. Rostrup-Nielsen, Methanation of CO over Nickel: Mechanism and Kinetics at High H₂/CO Ratios, *J. Phys. Chem. B*, 2005, **109**(6), 2432–2438.

148 G. Pina, C. Louis and M. A. Keane, Nickel particle size effects in catalytic hydrogenation and hydrodechlorination: phenolic transformations over nickel/silica, *Phys. Chem. Chem. Phys.*, 2003, **5**(9), 1924–1931.

149 A. Wolf, T. Turek and L. Mleczko, Structured Raney® Nickel Catalysts for Liquid-Phase Hydrogenation, *Chem. Eng. Technol.*, 2016, **39**(10), 1933–1938.

150 F. Devred, A. H. Gieske, N. Adkins, U. Dahlborg, C. M. Bao and M. Calvo-Dahlborg, *et al.*, Influence of phase composition and particle size of atomised Ni-Al alloy samples on the catalytic performance of Raney®-type nickel catalysts, *Appl. Catal., A*, 2009, **356**(2), 154–161.

151 Y. Tamaru, *Introductory Guide to Organonickel Chemistry, Modern Organonickel Chemistry*, Wiley-VCH Verlag GmbH & Co. KGaA, 2005, pp. 1–40.

152 M. J. Vaidya, S. M. Kulkarni and R. V. Chaudhari, Synthesis of p-Aminophenol by Catalytic Hydrogenation of p-Nitrophenol, *Org. Process Res. Dev.*, 2003, **7**(2), 202–208.

153 A. Thomas, A. Fischer, F. Goettmann, M. Antonietti, J.-O. Muller and R. Schlogl, *et al.*, Graphitic carbon nitride materials: variation of structure and morphology and their use as metal-free catalysts, *J. Mater. Chem.*, 2008, **18**(41), 4893–4908.

154 Z. Dong, C. Dong, Y. Liu, X. Le, Z. Jin and J. Ma, Hydrodechlorination and further hydrogenation of 4-chlorophenol to cyclohexanone in water over Pd nanoparticles modified N-doped mesoporous carbon microspheres, *Chem. Eng. J.*, 2015, **270**, 215–222.

155 W. Zuo, G. Yu and Z. Dong, A MOF-derived nickel based N-doped mesoporous carbon catalyst with high catalytic activity for the reduction of nitroarenes, *RSC Adv.*, 2016, **6**(14), 11749–11753.

156 S. Tang, S. Vongehr and X. Meng, Carbon Spheres with Controllable Silver Nanoparticle Doping, *J. Phys. Chem. C*, 2010, **114**(2), 977–982.

157 Y. Chi, Q. Yuan, Y. Li, J. Tu, L. Zhao and N. Li, *et al.*, Synthesis of Fe₃O₄@SiO₂-Ag magnetic nanocomposite based on small-sized and highly dispersed silver nanoparticles for catalytic reduction of 4-nitrophenol, *J. Colloid Interface Sci.*, 2012, **383**(1), 96–102.

158 N. Sahiner, H. Ozay, O. Ozay and N. Aktas, New catalytic route: Hydrogels as templates and reactors for *in situ* Ni nanoparticle synthesis and usage in the reduction of 2- and 4-nitrophenols, *Appl. Catal., A*, 2010, **385**(1–2), 201–207.

159 N. Sahiner, H. Ozay, O. Ozay and N. Aktas, A soft hydrogel reactor for cobalt nanoparticle preparation and use in the reduction of nitrophenols, *Appl. Catal., B*, 2010, **101**(1–2), 137–143.

160 H. Topsøe, B. S. Clausen and F. E. Massoth, Hydrotreating Catalysis, in *Catal.: Sci. Technol.*, ed. J. R. Anderson and M. Boudart, Berlin, Heidelberg, Springer Berlin Heidelberg, 1996, pp. 1–269.



161 P. D. C. Dietzel, B. Panella, M. Hirscher, R. Blom and H. Fjellvag, Hydrogen adsorption in a nickel based coordination polymer with open metal sites in the cylindrical cavities of the desolvated framework, *Chem. Commun.*, 2006, 959–961.

162 C. Larabi, P. K. Nielsen, S. Helveg, C. Thieuleux, F. B. Johansson and M. Brorson, *et al.*, Bulk Hydrodesulfurization Catalyst Obtained by Mo(CO)₆ Grafting on the Metal–Organic Framework Ni₂(2,5-dihydroxoterephthalate), *ACS Catal.*, 2012, 2(4), 695–700.

163 J. A. Rodríguez-Manzo, C. Pham-Huu and F. Banhart, Graphene Growth by a Metal-Catalyzed Solid-State Transformation of Amorphous Carbon, *ACS Nano*, 2011, 5(2), 1529–1534.

164 K. Takeishi and K.-I. Aika, Study of Raney[®] ruthenium catalyst for methanol synthesis, *J. Catal.*, 1992, 136(1), 252–257.

165 S. Dérien, F. Monnier and P. H. Dixneuf, Ruthenium-Catalyzed C–C Bond Formation, in *Ruthenium Catalysts and Fine Chemistry: -/-*, ed. C. Bruneau and P. H. Dixneuf, Berlin, Heidelberg, Springer Berlin Heidelberg, 2004, pp. 1–44.

166 S. Miao, Z. Liu, B. Han, J. Huang, Z. Sun and J. Zhang, *et al.*, Ru Nanoparticles Immobilized on Montmorillonite by Ionic Liquids: A Highly Efficient Heterogeneous Catalyst for the Hydrogenation of Benzene, *Angew. Chem., Int. Ed.*, 2006, 45(2), 266–269.

167 Y. Ma, Y. Huang, Y. Cheng, L. Wang and X. Li, Biosynthesized ruthenium nanoparticles supported on carbon nanotubes as efficient catalysts for hydrogenation of benzene to cyclohexane: An eco-friendly and economical bioreduction method, *Appl. Catal., A*, 2014, 484, 154–160.

168 T. Kamegawa, N. Suzuki and H. Yamashita, Design of macroporous TiO₂ thin film photocatalysts with enhanced photofunctional properties, *Energy Environ. Sci.*, 2011, 4(4), 1411–1416.

169 K. Khaletskaya, A. Pougin, R. Medishetty, C. Rösler, C. Wiktor and J. Strunk, *et al.*, Fabrication of Gold/Titania Photocatalyst for CO₂ Reduction Based on Pyrolytic Conversion of the Metal–Organic Framework NH₂-MIL-125(Ti) Loaded with Gold Nanoparticles, *Chem. Mater.*, 2015, 27(21), 7248–7257.

170 M. Dan-Hardi, C. Serre, T. Frot, L. Rozes, G. Maurin and C. Sanchez, *et al.*, A New Photoactive Crystalline Highly Porous Titanium(IV) Dicarboxylate, *J. Am. Chem. Soc.*, 2009, 131(31), 10857–10859.

171 S. N. Habisreutinger, L. Schmidt-Mende and J. K. Stolarczyk, Photocatalytic Reduction of CO₂ on TiO₂ and Other Semiconductors, *Angew. Chem., Int. Ed.*, 2013, 52(29), 7372–7408.

172 K. Li, X. An, K. H. Park, M. Khraisheh and J. Tang, A critical review of CO₂ photoconversion: Catalysts and reactors, *Catal. Today*, 2014, 224, 3–12.

173 B. Mei, A. Pougin and J. Strunk, Influence of photodeposited gold nanoparticles on the photocatalytic activity of titanate species in the reduction of CO₂ to hydrocarbons, *J. Catal.*, 2013, 306, 184–189.

174 A. Chica, A. Corma and M. E. Dómine, Catalytic oxidative desulfurization (ODS) of diesel fuel on a continuous fixed-bed reactor, *J. Catal.*, 2006, 242(2), 299–308.

175 N. D. McNamara, J. Kim and J. C. Hicks, Controlling the Pyrolysis Conditions of Microporous/Mesoporous MIL-125 To Synthesize Porous, Carbon-Supported Ti Catalysts with Targeted Ti Phases for the Oxidation of Dibenzothiophene, *Energy Fuels*, 2016, 30(1), 594–602.

176 J. Kim, G. T. Neumann, N. D. McNamara and J. C. Hicks, Exceptional control of carbon-supported transition metal nanoparticles using metal–organic frameworks, *J. Mater. Chem. A*, 2014, 2(34), 14014–14027.

177 S.-N. Kim, J. Kim, H.-Y. Kim, H.-Y. Cho and W.-S. Ahn, Adsorption/catalytic properties of MIL-125 and NH₂-MIL-125, *Catal. Today*, 2013, 204, 85–93.

178 D. Rehder, Vanadium. Its Role for Humans, in *Interrelations between Essential Metal Ions and Human Diseases*, ed. A. Sigel, H. Sigel and R. K. O. Sigel, Dordrecht, Springer Netherlands, 2013, pp. 139–169.

179 E. Hryha, E. Rutqvist and L. Nyborg, Stoichiometric vanadium oxides studied by XPS, *Surf. Interface Anal.*, 2012, 44(8), 1022–1025.

180 W. Xing, F. Meng and R. Yu, A new type of vanadium carbide V₅C₃ and its hardening by tuning Fermi energy, *Sci. Rep.*, 2016, 6, 21794.

181 G. Fang, W. Wu, C. Liu, D. D. Dionysiou, Y. Deng and D. Zhou, Activation of persulfate with vanadium species for PCBs degradation: A mechanistic study, *Appl. Catal., B*, 2017, 202, 1–11.

182 I. E. Wachs, Catalysis science of supported vanadium oxide catalysts, *Dalton Trans.*, 2013, 42(33), 11762–11769.

183 F. Meunier, P. Delporte, B. Heinrich, C. Bouchy, C. Crouzet and C. Pham-Huu, *et al.*, Synthesis and Characterization of High Specific Surface Area Vanadium Carbide; Application to Catalytic Oxidation, *J. Catal.*, 1997, 169(1), 33–44.

184 J. Ma, M. Wu, Y. Du, S. Chen, J. Ye and L. Jin, Low temperature synthesis of vanadium carbide (VC), *Mater. Lett.*, 2009, 63(11), 905–907.

185 I. J. Kang, N. A. Khan, E. Haque and S. H. Jhung, Chemical and Thermal Stability of Isotypic Metal–Organic Frameworks: Effect of Metal Ions, *Chem. – Eur. J.*, 2011, 17(23), 6437–6442.

186 J. Kim, N. D. McNamara and J. C. Hicks, Catalytic activity and stability of carbon supported V oxides and carbides synthesized via pyrolysis of MIL-47 (V), *Appl. Catal., A*, 2016, 517, 141–150.

187 F. Liu, Y. Yao, H. Zhang, Y. Kang, G. Yin and Z. Huang, *et al.*, Synthesis and characterization of vanadium carbide nanoparticles by thermal refluxing-derived precursors, *J. Mater. Sci.*, 2011, 46(11), 3693–3697.

188 X.-F. Wu, Non-Redox-Metal-Catalyzed Redox Reactions: Zinc Catalysts, *Chem. – Asian J.*, 2012, 7(11), 2502–2509.

189 G. Parravano and M. Boudart, Chemisorption and Catalysis on Oxide Semiconductors, in *Adv. Catal.*, ed. W. G. Frankenburg, V. I. Komarewsky and E. K. Rideal, Academic Press, 1955, vol. 7, pp. 47–74.



190 H. Li, M. Eddaoudi, M. O'Keeffe and O. M. Yaghi, Design and synthesis of an exceptionally stable and highly porous metal-organic framework, *Nature*, 1999, **402**(6759), 276–279.

191 B. Panella, M. Hirscher, H. Pütter and U. Müller, Hydrogen Adsorption in Metal–Organic Frameworks: Cu-MOFs and Zn-MOFs Compared, *Adv. Funct. Mater.*, 2006, **16**(4), 520–524.

192 S. S. Kaye, A. Dailly, O. M. Yaghi and J. R. Long, Impact of Preparation and Handling on the Hydrogen Storage Properties of $Zn_4O(1,4\text{-benzenedicarboxylate})_3$ (MOF-5), *J. Am. Chem. Soc.*, 2007, **129**(46), 14176–14177.

193 J. Li, S. Cheng, Q. Zhao, P. Long and J. Dong, Synthesis and hydrogen-storage behavior of metal–organic framework MOF-5, *Int. J. Hydrogen Energy*, 2009, **34**(3), 1377–1382.

194 D. Saha, Z. Bao, F. Jia and S. Deng, Adsorption of CO_2 , CH_4 , N_2O , and N_2 on MOF-5, MOF-177, and Zeolite 5A, *Environ. Sci. Technol.*, 2010, **44**(5), 1820–1826.

195 Z. Zhao, X. Ma, A. Kasik, Z. Li and Y. S. Lin, Gas Separation Properties of Metal Organic Framework (MOF-5) Membranes, *Ind. Eng. Chem. Res.*, 2013, **52**(3), 1102–1108.

196 N. T. S. Phan, K. K. A. Le and T. D. Phan, MOF-5 as an efficient heterogeneous catalyst for Friedel–Crafts alkylation reactions, *Appl. Catal., A*, 2010, **382**(2), 246–253.

197 W. Zhen, B. Li, G. Lu and J. Ma, Enhancing catalytic activity and stability for CO_2 methanation on $Ni@MOF-5$ via control of active species dispersion, *Chem. Commun.*, 2015, **51**(9), 1728–1731.

198 N. A. Rodriguez, R. Parra and M. A. Grela, Structural characterization, optical properties and photocatalytic activity of MOF-5 and its hydrolysis products: implications on their excitation mechanism, *RSC Adv.*, 2015, **5**(89), 73112–73118.

199 Y.-R. Lee, M.-S. Jang, H.-Y. Cho, H.-J. Kwon, S. Kim and W.-S. Ahn, ZIF-8: A comparison of synthesis methods, *Chem. Eng. J.*, 2015, **271**, 276–280.

200 L. H. Wee, N. Janssens, S. P. Sree, C. Wiktor, E. Gobechiya and R. A. Fischer, *et al.*, Local transformation of ZIF-8 powders and coatings into ZnO nanorods for photocatalytic application, *Nanoscale*, 2014, **6**(4), 2056–2060.

201 S. R. Venna and M. A. Carreon, Highly Permeable Zeolite Imidazolate Framework-8 Membranes for CO_2/CH_4 Separation, *J. Am. Chem. Soc.*, 2010, **132**(1), 76–78.

202 G. Lu, O. K. Farha, W. Zhang, F. Huo and J. T. Hupp, Engineering ZIF-8 Thin Films for Hybrid MOF-Based Devices, *Adv. Mater.*, 2012, **24**(29), 3970–3974.

203 C. Chizallet, S. Lazare, D. Bazer-Bachi, F. Bonnier, V. Lecocq and E. Soyer, *et al.*, Catalysis of Transesterification by a Nonfunctionalized Metal–Organic Framework: Acido-Basicity at the External Surface of ZIF-8 Probed by FTIR and ab Initio Calculations, *J. Am. Chem. Soc.*, 2010, **132**(35), 12365–12377.

204 O. M. Yaghi, M. O'Keeffe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, Reticular synthesis and the design of new materials, *Nature*, 2003, **423**(6941), 705–714.

205 T. Lee, H. Kim, W. Cho, D.-Y. Han, M. Ridwan and C. W. Yoon, *et al.*, Thermosensitive Structural Changes and Adsorption Properties of Zeolitic Imidazolate Framework-8 (ZIF-8), *J. Phys. Chem. C*, 2015, **119**(15), 8226–8237.

206 Y. Kimitsuka, E. Hosono, S. Ueno, H. Zhou and S. Fujihara, Fabrication of Porous Cubic Architecture of ZnO Using Zn-terephthalate MOFs with Characteristic Microstructures, *Inorg. Chem.*, 2013, **52**(24), 14028–14033.

207 R. Mohan, K. Krishnamoorthy and S.-J. Kim, Enhanced photocatalytic activity of Cu-doped ZnO nanorods, *Solid State Commun.*, 2012, **152**(5), 375–380.

208 D. Lin, H. Wu, R. Zhang and W. Pan, Enhanced Photocatalysis of Electrospun Ag– ZnO Heterostructured Nanofibers, *Chem. Mater.*, 2009, **21**(15), 3479–3484.

209 S. J. Yang, J. H. Im, T. Kim, K. Lee and C. R. Park, MOF-derived ZnO and $ZnO@C$ composites with high photocatalytic activity and adsorption capacity, *J. Hazard. Mater.*, 2011, **186**(1), 376–382.

210 L. Pan, T. Muhammad, L. Ma, Z.-F. Huang, S. Wang and L. Wang, *et al.*, MOF-derived C-doped ZnO prepared via a two-step calcination for efficient photocatalysis, *Appl. Catal., B*, 2016, **189**, 181–191.

211 Z.-Y. Yuan and B.-L. Su, Insights into hierarchically meso-macroporous structured materials, *J. Mater. Chem.*, 2006, **16**(7), 663–677.

212 A. Fukuoka, J.-I. Kimura, T. Oshio, Y. Sakamoto and M. Ichikawa, Preferential Oxidation of Carbon Monoxide Catalyzed by Platinum Nanoparticles in Mesoporous Silica, *J. Am. Chem. Soc.*, 2007, **129**(33), 10120–10125.

213 M. Haruta, Catalysis: Gold rush, *Nature*, 2005, **437**(7062), 1098–1099.

214 O. Pozdnyakova, D. Teschner, A. Woitsch, J. Kröhnert, B. Steinhauer and H. Sauer, *et al.*, Preferential CO oxidation in hydrogen (PROX) on ceria-supported catalysts, part I: Oxidation state and surface species on Pt/CeO_2 under reaction conditions, *J. Catal.*, 2006, **237**(1), 1–16.

215 B. Liu, S. Han, K. Tanaka, H. Shioyama and Q. Xu, Metal–Organic Framework (MOF) as a Precursor for Synthesis of Platinum Supporting Zinc Oxide Nanoparticles, *Bull. Chem. Soc. Jpn.*, 2009, **82**(8), 1052–1054.

216 D. Banham, F. Feng, K. Pei, S. Ye and V. Birss, Effect of carbon support nanostructure on the oxygen reduction activity of Pt/C catalysts, *J. Mater. Chem. A*, 2013, **1**(8), 2812–2820.

217 I. A. Khan, Y. Qian, A. Badshah, M. A. Nadeem and D. Zhao, Highly Porous Carbon Derived from MOF-5 as a Support of ORR Electrocatalysts for Fuel Cells, *ACS Appl. Mater. Interfaces*, 2016, **8**(27), 17268–17275.

218 W. Chen, J. Zhao, J. Y. Lee and Z. Liu, Microwave heated polyol synthesis of carbon nanotubes supported Pt nanoparticles for methanol electrooxidation, *Mater. Chem. Phys.*, 2005, **91**(1), 124–129.

219 I. A. Khan, A. Badshah, N. Haider, S. Ullah, D. H. Anjum and M. A. Nadeem, Porous carbon as electrode material in direct ethanol fuel cells (DEFCs) synthesized by the direct carbonization of MOF-5, *J. Solid State Electrochem.*, 2014, **18**(6), 1545–1555.



220 H. A. Gasteiger and N. M. Marković, Just a Dream—or Future Reality?, *Science*, 2009, **324**(5923), 48–49.

221 P. Strasser, S. Koh, T. Anniyev, J. Greeley, K. More and C. Yu, *et al.*, Lattice-strain control of the activity in dealloyed core-shell fuel cell catalysts, *Nat. Chem.*, 2010, **2**(6), 454–460.

222 X. Liu and L. Dai, Carbon-based metal-free catalysts, *Nat. Rev. Mater.*, 2016, 16064.

223 A. Ajaz, N. Fujiwara and Q. Xu, From Metal–Organic Framework to Nitrogen-Decorated Nanoporous Carbons: High CO₂ Uptake and Efficient Catalytic Oxygen Reduction, *J. Am. Chem. Soc.*, 2014, **136**(19), 6790–6793.

224 W. Zhang, Z.-Y. Wu, H.-L. Jiang and S.-H. Yu, Nanowire-Directed Templating Synthesis of Metal–Organic Framework Nanofibers and Their Derived Porous Doped Carbon Nanofibers for Enhanced Electrocatalysis, *J. Am. Chem. Soc.*, 2014, **136**(41), 14385–14388.

225 C. Feng, Y. Wang, S. Gao, N. Shang and C. Wang, Hydrogen generation at ambient conditions: AgPd bimetal supported on metal–organic framework derived porous carbon as an efficient synergistic catalyst, *Catal. Commun.*, 2016, **78**, 17–21.

226 K. Tanabe, Surface and catalytic properties of ZrO₂, *Mater. Chem. Phys.*, 1985, **13**(3), 347–364.

227 A. R. Puigdollers, F. Illas and G. Pacchioni, Structure and Properties of Zirconia Nanoparticles from Density Functional Theory Calculations, *J. Phys. Chem. C*, 2016, **120**(8), 4392–4402.

228 T. M. Miller and V. H. Grassian, Environmental Catalysis: Adsorption and Decomposition of Nitrous Oxide on Zirconia, *J. Am. Chem. Soc.*, 1995, **117**, 10969–10975.

229 T. Yokoyama, T. Setoyama, N. Fujita, M. Nakajima, T. Maki and K. Fujii, Novel direct hydrogenation process of aromatic carboxylic acids to the corresponding aldehydes with zirconia catalyst, *Appl. Catal., A*, 1992, **88**(2), 149–161.

230 I. Ferino, M. F. Casula, A. Corrias, M. G. Cutrufello, R. Monaci and G. Paschina, 4-Methylpentan-2-ol dehydration over zirconia catalysts prepared by sol-gel, *Phys. Chem. Chem. Phys.*, 2000, **2**(8), 1847–1854.

231 Y. Nakano, T. Iizuka, H. Hattori and K. Tanabe, Surface properties of zirconium oxide and its catalytic activity for isomerization of 1-butene, *J. Catal.*, 1979, **57**(1), 1–10.

232 M.-Y. He and J. G. Ekerdt, Methanol formation on zirconium dioxide, *J. Catal.*, 1984, **90**(1), 17–23.

233 A. E. Platero-Prats, A. Mavrandakis, L. C. Gallington, Y. Liu, J. T. Hupp and O. K. Farha, *et al.*, Structural Transitions of the Metal-Oxide Nodes within Metal–Organic Frameworks: On the Local Structures of NU-1000 and UiO-66, *J. Am. Chem. Soc.*, 2016, **138**(12), 4178–4185.

234 D. Yang, S. O. Odoh, T. C. Wang, O. K. Farha, J. T. Hupp and C. J. Cramer, *et al.*, Metal–Organic Framework Nodes as Nearly Ideal Supports for Molecular Catalysts: NU-1000- and UiO-66-Supported Iridium Complexes, *J. Am. Chem. Soc.*, 2015, **137**(23), 7391–7396.

235 J. E. Mondloch, M. J. Katz, W. C. Isley III, P. Ghosh, P. Liao and W. Bury, *et al.*, Destruction of chemical warfare agents using metal–organic frameworks, *Nat. Mater.*, 2015, **14**(5), 512–516.

236 W. Zhou, E. I. Ross-Medgaarden, W. V. Knowles, M. S. Wong, I. E. Wachs and C. J. Kiely, Identification of active Zr-WO_x clusters on a ZrO₂ support for solid acid catalysts, *Nat. Chem.*, 2009, **1**(9), 722–728.

237 G. Larsen, E. Lotero, L. M. Petkovic and D. S. Shobe, Alcohol Dehydration Reactions over Tungstated Zirconia Catalysts, *J. Catal.*, 1997, **169**(1), 67–75.

238 F. Di Gregorio and V. Keller, Activation and isomerization of hydrocarbons over WO₃/ZrO₂ catalysts: I. Preparation, characterization, and X-ray photoelectron spectroscopy studies, *J. Catal.*, 2004, **225**(1), 45–55.

239 P. Wang, J. Feng, Y. Zhao, S. Wang and J. Liu, MOF-Derived Tungstated Zirconia as Strong Solid Acids toward High Catalytic Performance for Acetalization, *ACS Appl. Mater. Interfaces*, 2016, **8**(36), 23755–23762.

240 P. Zhang, F. Sun, Z. Xiang, Z. Shen, J. Yun and D. Cao, ZIF-derived *in situ* nitrogen-doped porous carbons as efficient metal-free electrocatalysts for oxygen reduction reaction, *Energy Environ. Sci.*, 2014, **7**(1), 442–450.

241 Z.-W. Liu, F. Peng, H.-J. Wang, H. Yu, W.-X. Zheng and J. Yang, Phosphorus-Doped Graphite Layers with High Electrocatalytic Activity for the O₂ Reduction in an Alkaline Medium, *Angew. Chem., Int. Ed.*, 2011, **50**(14), 3257–3261.

242 Y. Bai, Y. Dou, L.-H. Xie, W. Rutledge, J.-R. Li and H.-C. Zhou, Zr-based metal–organic frameworks: design, synthesis, structure, and applications, *Chem. Soc. Rev.*, 2016, **45**(8), 2327–2367.

243 Y. Fu, Y. Huang, Z. Xiang, G. Liu and D. Cao, Phosphorous–Nitrogen-Codoped Carbon Materials Derived from Metal–Organic Frameworks as Efficient Electrocatalysts for Oxygen Reduction Reactions, *Eur. J. Inorg. Chem.*, 2016, 2100–2105.

244 W. Yu, M. D. Porosoff and J. G. Chen, Review of Pt-Based Bimetallic Catalysis: From Model Surfaces to Supported Catalysts, *Chem. Rev.*, 2012, **112**(11), 5780–5817.

245 Y. Yang, Y. Zhang, C. J. Sun, X. Li, W. Zhang and X. Ma, *et al.*, Heterobimetallic Metal–Organic Framework as a Precursor to Prepare a Nickel/Nanoporous Carbon Composite Catalyst for 4-Nitrophenol Reduction, *ChemCatChem*, 2014, **6**(11), 3084–3090.

246 A. D. Burrows, Mixed-component metal–organic frameworks (MC-MOFs), enhancing functionality through solid solution formation and surface modifications, *CrystEngComm*, 2011, **13**(11), 3623–3642.

247 B. You, N. Jiang, M. Sheng, W. S. Drisdell, J. Yano and Y. Sun, Bimetal–Organic Framework Self-Adjusted Synthesis of Support-Free Nonprecious Electrocatalysts for Efficient Oxygen Reduction, *ACS Catal.*, 2015, **5**(12), 7068–7076.

248 X. Wang, X. Fan, H. Lin, H. Fu, T. Wang and J. Zheng, *et al.*, An efficient Co–N–C oxygen reduction catalyst with highly dispersed Co sites derived from a ZnCo bimetallic zeolitic imidazolate framework, *RSC Adv.*, 2016, **6**(44), 37965–37973.



249 W. Zhou, J. Zhou, Y. Zhou, J. Lu, K. Zhou and L. Yang, *et al.*, N-Doped Carbon-Wrapped Cobalt Nanoparticles on N-Doped Graphene Nanosheets for High-Efficiency Hydrogen Production, *Chem. Mater.*, 2015, **27**(6), 2026–2032.

250 J. Lu, W. Zhou, L. Wang, J. Jia, Y. Ke and L. Yang, *et al.*, Core–Shell Nanocomposites Based on Gold Nanoparticle@Zinc–Iron-Embedded Porous Carbons Derived from Metal–Organic Frameworks as Efficient Dual Catalysts for Oxygen Reduction and Hydrogen Evolution Reactions, *ACS Catal.*, 2016, **6**(2), 1045–1053.

251 L. He, Y. Liu, J. Liu, Y. Xiong, J. Zheng and Y. Liu, *et al.*, Core–Shell Noble-Metal@Metal–Organic-Framework Nanoparticles with Highly Selective Sensing Property, *Angew. Chem., Int. Ed.*, 2013, **52**(13), 3741–3745.

252 Q. Xu, Z. Guo, M. Zhang, Z. Hu, Y. Qian and D. Zhao, Highly efficient photocatalysts by pyrolyzing a Zn-Ti heterometallic metal–organic framework, *CrystEngComm*, 2016, **18**(22), 4046–4052.

253 Z. Guo, J. K. Cheng, Z. Hu, M. Zhang, Q. Xu and Z. Kang, *et al.*, Metal–organic frameworks (MOFs) as precursors towards TiO_x/C composites for photodegradation of organic dye, *RSC Adv.*, 2014, **4**(65), 34221–34225.

254 K. Hong, W. Bak and H. Chun, Unique Coordination-Based Heterometallic Approach for the Stoichiometric Inclusion of High-Valent Metal Ions in a Porous Metal–Organic Framework, *Inorg. Chem.*, 2013, **52**(10), 5645–5647.

255 K.-L. Wu, X.-W. Wei, X.-M. Zhou, D.-H. Wu, X.-W. Liu and Y. Ye, *et al.*, NiCo₂ Alloys: Controllable Synthesis, Magnetic Properties, and Catalytic Applications in Reduction of 4-Nitrophenol, *J. Phys. Chem. C*, 2011, **115**(33), 16268–16274.

256 Z. Zhu, X. Guo, S. Wu, R. Zhang, J. Wang and L. Li, Preparation of Nickel Nanoparticles in Spherical Polyelectrolyte Brush Nanoreactor and Their Catalytic Activity, *Ind. Eng. Chem. Res.*, 2011, **50**(24), 13848–13853.

257 Y. Yang, Y. Ren, C. Sun and S. Hao, Facile route fabrication of nickel based mesoporous carbons with high catalytic performance towards 4-nitrophenol reduction, *Green Chem.*, 2014, **16**(4), 2273–2280.

258 L. Zheng, X. Li, W. Du, D. Shi, W. Ning and X. Lu, *et al.*, Metal–organic framework derived Cu/ZnO catalysts for continuous hydrogenolysis of glycerol, *Appl. Catal., B*, 2017, **203**, 146–153.

259 A. Kong, Q. Lin, C. Mao, X. Bu and P. Feng, Efficient oxygen reduction by nanocomposites of heterometallic carbide and nitrogen-enriched carbon derived from the cobalt-encapsulated indium-MOF, *Chem. Commun.*, 2014, **50**(98), 15619–15622.

260 J. Long, K. Shen, L. Chen and Y. Li, Multimetal-MOF-derived transition metal alloy NPs embedded in an N-doped carbon matrix: highly active catalysts for hydrogenation reactions, *J. Mater. Chem. A*, 2016, **4**(26), 10254–10262.

261 W. Weisweiler, N. Subramanian and B. Terwiesch, Catalytic influence of metal melts on the graphitization of monolithic glasslike carbon, *Carbon*, 1971, **9**(6), 755–761.

