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# Barium titanate photocatalysts with silver–manganese dual cocatalyst for carbon dioxide reduction with water†

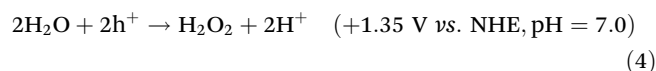
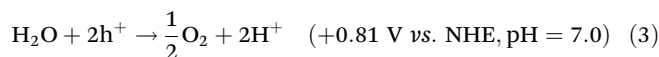
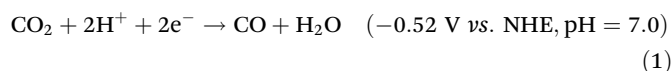
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Titanate photocatalysts with suitable cocatalysts are promising candidates for photocatalytic CO<sub>2</sub> reduction, with the use of water as an electron donor. Here, several barium titanates with various compositions were examined, and BaTi<sub>4</sub>O<sub>9</sub> (BT4) was found to be the best photocatalyst with the assistance of an Ag cocatalyst for photocatalytic CO<sub>2</sub> reduction to form CO. The photocatalytic activity was further enhanced by the use of MnO<sub>x</sub> as an additional cocatalyst to construct an Ag–MnO<sub>x</sub>/BT4 photocatalyst, where Ag and MnO<sub>x</sub> were selectively deposited at different facets on BT4 crystal and functioned as active sites for CO<sub>2</sub> reduction and water oxidation, respectively. As for the oxidative products from water, molecular oxygen (O<sub>2</sub>) and hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>) were obtained.

## 1. Introduction

Excess consumption of natural fossil resources has led to an increase in atmospheric greenhouse gases. Particularly, the high concentration of carbon dioxide (CO<sub>2</sub>) causing climate change has become a global issue,<sup>1</sup> requiring the development of new technology to convert CO<sub>2</sub> to valuable compounds without using fossil fuels.<sup>2</sup> Artificial photosynthesis using water as a reductant is one such methodology to provide a sustainable global carbon cycle, where CO<sub>2</sub> can be converted with solar energy into basic chemicals such as carbon monoxide (CO), formaldehyde, methanol, and methane.<sup>3,4</sup> Among these possible products, CO is a useful compound because it is easily separated from the aqueous reaction solution, and it is well known as a component of the synthesis gas valuably utilized for the Fischer–Tropsch reaction.<sup>5,6</sup> However, the efficiency of the energy conversion in photocatalysis remains far from satisfactory.

In this photocatalytic reaction, CO<sub>2</sub> is reduced through a reaction with protons and electrons to form CO (eqn (1)), which is competitive with hydrogen (H<sub>2</sub>) formation (eqn (2)). Water is oxidized to O<sub>2</sub> (eqn (3)) or H<sub>2</sub>O<sub>2</sub> (eqn (4)) and reproduced in the reductive reaction (eqn (1)).



Because the redox potential of H<sup>+</sup>/H<sub>2</sub> is more positive than that of CO/CO<sub>2</sub>,<sup>7</sup> it is difficult to selectively activate CO<sub>2</sub> molecules in aqueous reaction conditions. Kudo and coworkers<sup>8</sup> reported that CO<sub>2</sub> can be predominantly reduced into CO even in water when silver nanoparticles (Ag NPs) serve as an efficient cocatalyst on the BaLa<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub> photocatalyst in the photocatalytic reduction of CO<sub>2</sub>, with water as an electron donor. Following this, many photocatalysts have been investigated, such as Ag/La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>,<sup>9</sup> Ag/Sr<sub>2</sub>KTa<sub>5</sub>O<sub>15</sub>,<sup>10</sup> Ag/KCaSrTa<sub>5</sub>O<sub>15</sub>,<sup>11</sup> Ag/ZnGa<sub>2</sub>O<sub>4</sub>,<sup>12</sup> Ag/Ga<sub>2</sub>O<sub>3</sub>,<sup>13</sup> Ag/ZnTa<sub>2</sub>O<sub>6</sub>,<sup>14</sup> Ag/CaTiO<sub>3</sub>,<sup>15</sup> and Ag/K<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub>.<sup>16</sup> However, it is imperative to develop new photocatalysts with higher energy conversion efficiency and reaction selectivity.

Plentiful works demonstrate that barium titanate materials, which are ferroelectric, inexpensive, and environmentally friendly,<sup>17</sup> are promising candidates for the photodegradation of organic compounds,<sup>18–20</sup> N<sub>2</sub> reduction,<sup>21</sup> and water splitting.<sup>22–24</sup> It has been reported that BaTiO<sub>3</sub> is an efficient photocatalyst for the CO<sub>2</sub> photoreduction of CH<sub>4</sub> and CO.<sup>25</sup> Thus, other barium titanates with different compositions are also attractive to be examined for this photocatalytic reaction. Among these barium titanates, BaTi<sub>4</sub>O<sub>9</sub> (BT4) with a pentago-

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nal-prism tunnel structure, which brings out a significantly moderate distortion of  $\text{TiO}_6$  octahedra possibly leading to enhancement of the field polarization and efficient production of photoexcited charges,<sup>26,27</sup> is a satisfactory photocatalyst for  $\text{CO}_2$  reduction.<sup>28</sup> Thus, other barium titanates with different compositions may also be well-suited for this photocatalytic reaction.

Charge separation in photocatalyst crystals covered with facets is a beneficial and crucial method to increase the photocatalytic activity with less electron-hole recombination because photogenerated electrons and holes can migrate to respective facets to promote the reductive and oxidative reactions on the separated surface.<sup>29</sup> For instance, Li *et al.* demonstrated that (010) facets prefer a reductive reaction, whereas (110) facets favour an oxidative reaction by loading metals (Au, Ag, Pt) and metal oxides ( $\text{MnO}_x$ ,  $\text{PbO}_2$ ), respectively, on the surface of monoclinic  $\text{BiVO}_4$  under light irradiation.<sup>30</sup> It was verified that a cocatalyst deposited on reductive or oxidative facets selectively results in higher catalytic activity than a randomly dispersed cocatalyst.<sup>31</sup> Our previous studies revealed that a combination of Ag- $\text{MnO}_x$  cocatalysts selectively loaded on a hexagonal rod-like  $\text{K}_2\text{Ti}_6\text{O}_{13}$  crystal photocatalyst with clear facets showed a high and selective activity for  $\text{CO}_2$  reduction,<sup>32</sup> while the same dual cocatalyst loaded on a  $\text{CaTiO}_3$  photocatalyst converted  $\text{CO}_2$  and  $\text{H}_2\text{O}$  to CO and  $\text{H}_2\text{O}_2$ ,<sup>33</sup> showing that the reaction selectivity depends on the combination of photocatalyst and cocatalyst.

In the present work, a series of barium titanates with different compositions was synthesized and loaded with Ag NP cocatalyst for subsequent examination of photocatalytic  $\text{CO}_2$  reduction with water. A higher CO formation rate with greater selectivity was obtained with an Ag/BT4 sample as compared to others. Thus, the preparation conditions for the BT4 photocatalyst were optimized, and an Ag- $\text{MnO}_2$  dual cocatalyst was tested for BT4, which further enhanced the photocatalytic activity two-fold.

## 2. Experimental section

### 2.1 Preparation of samples

BT4 was fabricated by a solid-state reaction method according to a previous report.<sup>28</sup> Briefly,  $\text{BaCO}_3$  (Rare Metallic Co., Ltd) and  $\text{TiO}_2$  (rutile, Kojundo Chemical Lab. Co., Ltd) were mixed with a molar ratio of Ba:Ti = 1:4 for 20 min in an alumina mortar in the presence of ethanol. The obtained powder was placed into an alumina crucible and heated in a muffle furnace, where the temperature was increased by a rate of 10 K  $\text{min}^{-1}$  from room temperature to 1173 K, 1273 K, or 1373 K (typically 1273 K) and maintained for 5, 10, 20, or 50 hours (typically 20 h), followed by cooling to room temperature. The BT4 samples calcined for  $x$  h at  $y$  K are referred to as BT4( $x,y$ ).

$\text{BaTi}_n\text{O}_{2n+1}$  ( $n = 1, 5$ ) and  $\text{Ba}_{2(n-1)}\text{Ti}_{4n+1}\text{O}_{10n}$  ( $n = 2, 3, 4$ ) were prepared by a solid state reaction method,<sup>24,34,35</sup> while  $\text{BaTi}_2\text{O}_5$  was synthesized by a flux method,<sup>36</sup> as described in the ESI.†

The Ag cocatalyst was loaded on these barium titanate photocatalysts by a photodeposition (PD) method. The single Ag and single  $\text{MnO}_x$  cocatalyst, or Ag and  $\text{MnO}_x$  dual cocatalyst, were employed to modify the surface of the as-prepared BT4 by the PD or a simultaneous photodeposition (SPD) method, respectively. In these photodeposition methods, 2 g of BT4 sample was dispersed in 750 mL of ion-exchange water in an inner-irradiation reactor (Fig. S1†), and a certain amount of aqueous solution of  $\text{Ag}(\text{NO}_3)_3$  (0.5 M),  $\text{Mn}(\text{NO}_3)_2$  (0.1 M), or  $\text{Ag}(\text{NO}_3)_3$  (0.5 M) and  $\text{Mn}(\text{NO}_3)_2$  (0.1 M) was solely or simultaneously added to the suspension. After subjection to flowing argon with bubbling at a rate of 30 mL  $\text{min}^{-1}$  for 1.5 h in dark, the suspension was photoirradiated for 3 h using a high-pressure 100 W Hg lamp with Ar bubbling, where the lamp in a quartz cooling jacket connected to a water-cooling system was located at the centre of the reactor (Fig. S1†). Subsequently, the powder in the solution was filtered with suction and dried in an oven at 353 K overnight. An impregnation method (IMP) was also employed for Ag deposition, as shown in the ESI.†

### 2.2 Characterization

The crystal structure of each sample was determined by X-ray diffraction (XRD) with a Lab X XRD-6000 diffractometer (Shimadzu). Scanning electron microscopy (SEM) images were recorded by a field-emission scanning electron microscope (SU-8220, Hitachi, Japan), where elemental mappings were obtained by energy-dispersive X-ray spectroscopy (EDX, 15.0 kV). Diffuse reflectance (DR) UV-Vis spectroscopy of the prepared samples was performed using a V-570 (JASCO) instrument at room temperature with  $\text{BaSO}_4$  as a reference. The compositions of cocatalysts in the samples were evaluated by X-ray fluorescence (XRF) analysis with an EDX-8000 spectrometer (Shimadzu) in accordance with each calibration curve obtained by reference samples prepared by an impregnation method. The Brunauer-Emmett-Teller (BET) specific area was evaluated with a Monosorb (Quantachrome) at 77 K. Ag and Mn K-edge X-ray absorption spectra (XAS) were recorded at NW-10A in transmission mode and BL-9A in fluorescence mode of the Photon Factory (KEK, Tsukuba, Japan), respectively.

### 2.3 Photocatalytic reaction

Photocatalytic reaction tests were carried out in the inner-irradiation vessel (Fig. S1†). The photocatalyst of 1.0 g was placed into the reactor with 750 mL ion-exchanged water containing 0.5 M  $\text{NaHCO}_3$  as a buffer, and the solution was magnetically stirred without irradiation under a  $\text{CO}_2$  flow with a flow rate of 30 mL  $\text{min}^{-1}$  for 1.5 h to purge the reactor. After initiating the photoirradiation with a 100 W Hg lamp (approximately 8.2 mW  $\text{cm}^{-2}$  measured at  $360 \pm 10$  nm in wavelength) located at the centre of the reactor, the temperature was maintained at 283 K in the reactor with a flow of cool water. The products in the outflowing gas from the reactor were sampled at each scheduled time and analysed by online gas chromatography (Shimadzu, GC-8A, TCD, Shincarbon ST column, argon carrier). Typically, the data recorded after 3 h from the start of photoirradiation were employed for discussion. The primary



gaseous products were limited to H<sub>2</sub>, O<sub>2</sub> and CO, while the compound analysed from the liquid phase was limited to H<sub>2</sub>O<sub>2</sub>.

Because there is no other reductive product in the present work, the selectivity toward CO evolution,  $S_{CO}$  was calculated by eqn (5). The ratio of consumed electrons and holes,  $R(e^-/h^+)$ , in the process of CO<sub>2</sub> photoreduction is defined by eqn (6).

$$S_{CO}(\%) = 100 \times R_{CO} / (R_{H_2} + R_{CO}) \quad (5)$$

$$R(e^-/h^+) = (R_{CO} + R_{H_2}) / (2R_{O_2} + R_{H_2O_2}) \quad (6)$$

The formation rates of CO, H<sub>2</sub>, O<sub>2</sub>, and H<sub>2</sub>O<sub>2</sub> are referred to as  $R_{CO}$ ,  $R_{H_2}$ ,  $R_{O_2}$ , and  $R_{H_2O_2}$ , respectively. It is worthy to know that the value of  $R(e^-/h^+)$  should be unity because of the same amount of photogenerated electrons and holes.

### 3. Results and discussion

#### 3.1. Various barium titanates

Fig. S2† shows the XRD patterns of the barium titanate samples, BaTi<sub>n</sub>O<sub>2n+1</sub> ( $n = 1, 2, 4,$  and  $5$ ) and Ba<sub>2(n-1)</sub>Ti<sub>4n+1</sub>O<sub>10n</sub> ( $n = 2, 3,$  and  $4$ ). According to previous studies, most of these barium titanates including BaTiO<sub>3</sub> (BT), Ba<sub>6</sub>Ti<sub>17</sub>O<sub>40</sub> (B6T17), Ba<sub>4</sub>Ti<sub>13</sub>O<sub>30</sub> (B4T13), and Ba<sub>2</sub>Ti<sub>9</sub>O<sub>20</sub> (B2T9) can be successfully synthesized without an impurity phase,<sup>24,28,34</sup> except for BaTi<sub>4</sub>O<sub>9</sub> (BT4), BaTi<sub>2</sub>O<sub>5</sub> (BT2), and BaTi<sub>5</sub>O<sub>11</sub> (BT5). It was also noted that there was a tiny impurity phase of BaTiO<sub>3</sub> in the BT4 sample. The prepared BT2 sample included a certain amount of BaTiO<sub>3</sub> phase as an impurity.

In the phase diagram, the BaTi<sub>2</sub>O<sub>5</sub> phase can be found in a very narrow temperature window, which demonstrates its thermodynamically stability between 1220 and 1230 °C (1493–1503 K).<sup>37</sup> This indicates the requirement for further precise control of conditions to yield the pure BT2 phase due to the limited thermodynamic stability. Under the present conditions, the BT5 sample could not be successfully synthesized, and the obtained material was a mixture of the BaTi<sub>4</sub>O<sub>9</sub> phase and others, such as a trace amount of BaTiO<sub>3</sub> phase. A possible reason for this may be inhomogeneous mixing or uneven diffusion of the elements in the solid-state reaction due to insufficient heating time.<sup>38</sup>

Fig. 1 shows the DR UV–Vis spectra of the prepared barium titanate samples, which provides evidence that these materials can be excited by UV light.

After loading Ag cocatalyst of 1 wt% on these samples (except for the BT5 sample), their performances for the photocatalytic conversion of CO<sub>2</sub> with water were evaluated. Fig. 2 shows the formation rates of H<sub>2</sub>, O<sub>2</sub>, and CO and selectivity for CO ( $S_{CO}$ ) for the prepared photocatalysts. The BT, BT2, B6T17, and B2T9 samples loaded with the Ag cocatalyst showed trace amounts of H<sub>2</sub> and O<sub>2</sub> evolution. There was CO formation with low CO selectivity ( $S_{CO} = 39\%$ ) for the Ag/B4T13 sample. Ag/BT4 showed a prominent CO production rate of 4.5 μmol h<sup>-1</sup> among these Ag-loaded samples. This is the first report exam-

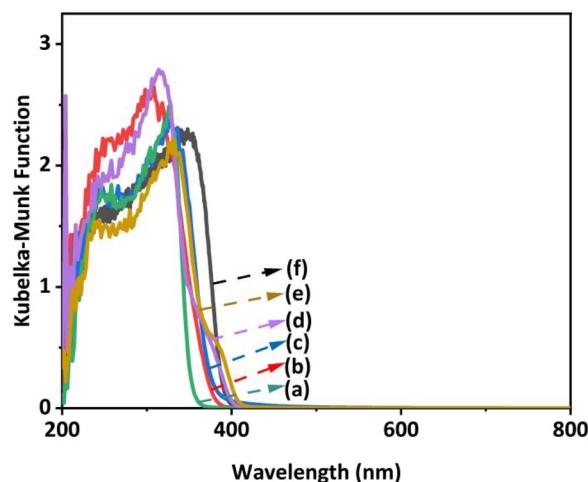


Fig. 1 DR UV–Vis spectra of the prepared samples: (a) B4T13, (b) BT2, (c) B6T17, (d) BT4, (e) B2T9, and (f) BT.

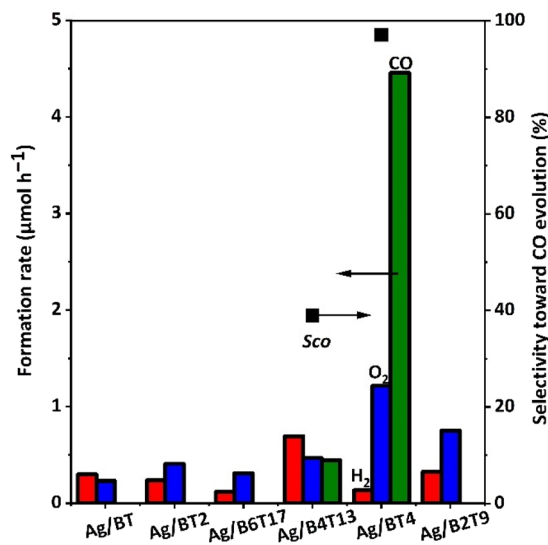


Fig. 2 Formation rates of H<sub>2</sub> (red), O<sub>2</sub> (blue), and CO (green), and selectivity for CO evolution ( $S_{CO}$ , square) in the photocatalytic reaction test for CO<sub>2</sub> reduction with water under photoirradiation over the barium titanate samples with 1 wt% Ag cocatalyst.

ining these six types of barium titanates for photocatalytic CO<sub>2</sub> reduction with water, and only the Ag/BT4 sample exhibited selective photocatalytic activity for CO<sub>2</sub> reduction to CO, with high selectivity of  $S_{CO} = 97\%$ .

The origin of the high photocatalytic performance of the Ag/BT4 sample is discussed here. From the DR UV–Vis spectra of the bare barium titanate samples (except for BT5) (Fig. 1), the bandgap was estimated using a Tauc plot<sup>39</sup> for each sample, as shown in Fig. S3† and listed in Table S1.† Assuming the tops of the valence bands of these barium titanates are near to each other,<sup>40</sup> the bottom of the conduction band was tentatively estimated (Table S1†). When compared to the redox potential of CO<sub>2</sub>/CO (Fig. S4†), the photoexcited elec-



trons in the valence band of the BT4 and B4T13 samples were expected to be able to reduce CO<sub>2</sub> to CO, and the activity of the BT4 photocatalyst was expected to be higher than that of B4T13, due to the greater negative potential of the excited electrons, which was consistent with the results (Fig. 2).

From a structural point of view, the structure of BT4 is unique, consisting of a pentagonal prism tunnel with structural distortion that provides a unique internal electric field that might reduce the recombination of the photoexcited electrons and holes.<sup>24</sup> Lin *et al.* proposed a correlation between the packing factor (PF) and photocatalytic activity among antimonate photocatalysts,<sup>41</sup> *i.e.*, the crystal of lower PF provides higher photocatalytic activity. The lowest PF among these six barium titanates is for BT4 (Table S2†), which exhibited the highest activity. At present, the origin of the high photocatalytic activity of the Ag/BT4 photocatalyst is unknown, and requires further detailed study.

Because the BT4 sample showed the highest photocatalytic activity among these obtained barium titanate photocatalysts, the study was continued with BT4. This does not preclude the possibility that the remaining barium titanates with other compositions may exhibit a high photocatalytic activity for CO<sub>2</sub> reduction after further optimization.

### 3.2. Preparation conditions for BaTi<sub>4</sub>O<sub>9</sub>

The optimization of BT4 was conducted by altering the calcination conditions. Fig. S5† shows the XRD patterns of the prepared BT4(*x*,1273) samples under different holding times of 5–50 h, at 1273 K. The primary diffractions of the prepared samples are identical to that of the BaTi<sub>4</sub>O<sub>9</sub> phase (ICSD #49575), and the diffraction intensities were similar to each other, which indicates that no difference was found according to XRD. It is noted that a trace amount of BaTiO<sub>3</sub> as an impurity phase was detected in all the BT4(*x*,1273) samples. A possible reason for this phenomenon may be the different melting temperatures, *i.e.*, the mixture of BaO + TiO<sub>2</sub> melts more easily than that of BaO + 4TiO<sub>2</sub> during calcination.<sup>42</sup> In addition, most of the samples showed tiny diffractions from rutile TiO<sub>2</sub>, an impurity phase, while no diffractions from BaCO<sub>3</sub> were found, meaning that unreacted TiO<sub>2</sub> remained as a crystallite, while unreacted Ba species possibly existed as BaCO<sub>3</sub> with very small crystallites or an amorphous phase.

Fig. 3 shows SEM images of the BT4(*x*,1273) samples prepared with different holding times. When the holding times were short, such as 5 h and 10 h, irregularly shaped BT4 particles agglomerated, and the characteristic polyhedral morphology could not form because of the insufficient calcination time. With increasing holding time such as 20 h and 50 h, the edge of the polyhedral BT4 crystal became clearer, suggesting that the facet junctions of BT4 clearly appeared, and the crystal sizes were larger by the order of several hundreds of nanometres.

Fig. S6† presents the XRD patterns of the BT4(20,*y*) samples prepared at different temperatures under the same holding time of 20 h. The diffraction peaks of all the samples are consistent with the BaTi<sub>4</sub>O<sub>9</sub> phase, although the calcination temp-

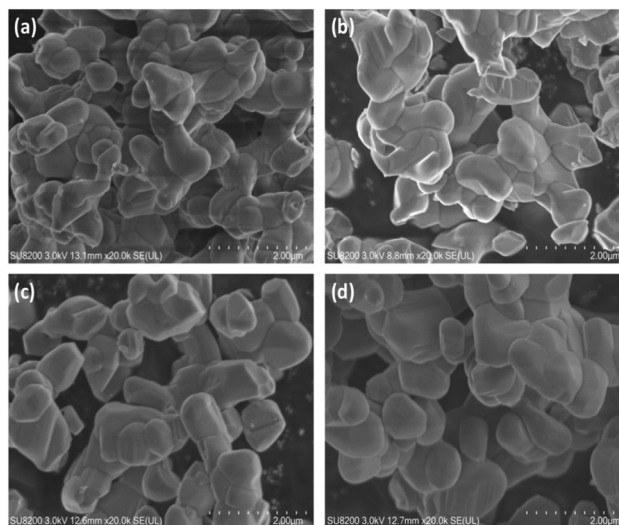


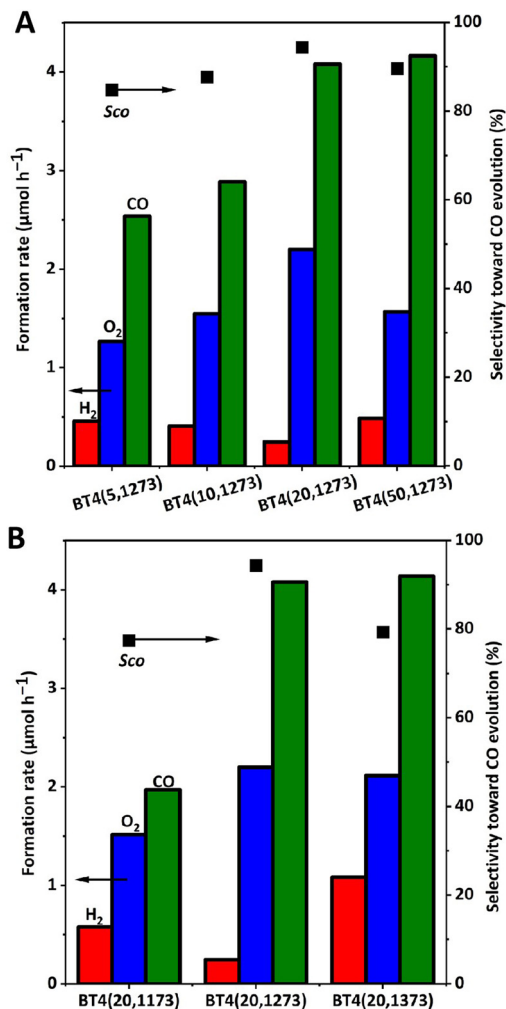
Fig. 3 SEM images of the BT4(*x*,1273) samples prepared by different holding times (*x*) of (a) 5, (b) 10, (c) 20, and (d) 50 h. Calcination was carried out at 1273 K.

erature is different. However, the impurity phase of BaTiO<sub>3</sub> did not disappear, and thus, the parameters of holding time or calcination temperature are not crucial for removing the impurity phase in the BT4(20,*y*) samples. The BT4(20,1173) sample showed diffractions with relatively low intensity compared with other BT4(20,*y*) samples calcined at higher temperatures. As shown in the SEM image (Fig. S7†), the BT4(20,1173) sample contained some smaller particles that might be incompletely crystallised. With increasing calcination temperature, the size of the BT4 particles increased from a hundred nanometres to one micrometre, and the shape became more regular.

Fig. 4 shows the results of the photocatalytic reaction tests for these BT4(*x*,*y*) samples with 1 wt% of Ag cocatalyst. Among the BT4(*x*,1273) samples prepared at 1273 K with different holding times (Fig. 4A), a relatively high CO formation rate was obtained for the BT4(20,1273) sample, with the highest selectivity for CO compared with the samples calcined under other holding times. Particularly, the photocatalytic performance was enhanced 1.6 times for the BT4(20,1273) sample compared with the BT4(5,1273) sample. Among the BT4(20,*y*) samples calcined at various temperatures for 20 h, the BT4(20,1273) sample exhibited the highest formation rate with the highest selectivity for CO evolution of 94.3% (Fig. 4B). Therefore, the optimum synthesis conditions were determined to be 1273 K for 20 h, and further study was conducted with the BT4 sample synthesized under these conditions, *i.e.*, the BT4(20,1273) sample.

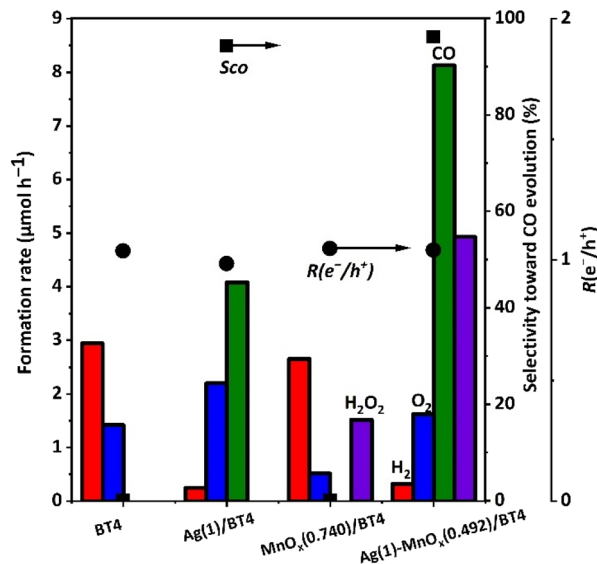
In many cases, specific surface area (SSA) is an important factor that affects the quantity of catalytic active sites on the surface of a photocatalyst, leading to different photocatalytic activities. The SSAs of these BT4(*x*,*y*) samples are listed in Table S3.† The results suggest that there is higher photocatalytic activity for the BT4 samples with smaller specific





**Fig. 4** Formation rates of H<sub>2</sub> (red), O<sub>2</sub> (blue), and CO (green), and selectivity for CO evolution ( $S_{CO}$ , square) in the photocatalytic reaction test for CO<sub>2</sub> photoreduction, with water as an electron donor under irradiation over the (A) BT4(*x*,1273) samples calcined for different holding times (*x*) at 1273 K and the (B) BT4(20,*y*) samples calcined at different calcination temperatures (*y*) for 20 h. The Ag loading amount was 1 wt%, as confirmed by XRF.

surface areas, such as Ag/BT4(20,1273) and Ag/BT4(50,1273) (Table S3,† entries 3 and 4), which is in agreement with the case of K<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub>, as reported in our previous work.<sup>43–45</sup> Larger particles with less SSA would have the advantage of a large cross-sectional area for photoabsorption, and would exhibit higher photocatalytic activity in photocatalysis using multielectrons such as CO<sub>2</sub> reduction with water oxidation (eqn (1) and (3)).<sup>43</sup> The other possible reasons for the difference in photocatalytic activity would be the high crystallinity and clear morphology with facets, as supported by XRD and SEM images, which will be discussed later. However, the BT4(20,1373) sample with the smallest SSA exhibited low photocatalytic activity (Table S3,† entry 6), which showed that the SSA is not the sufficient condition, but one of the requirements.



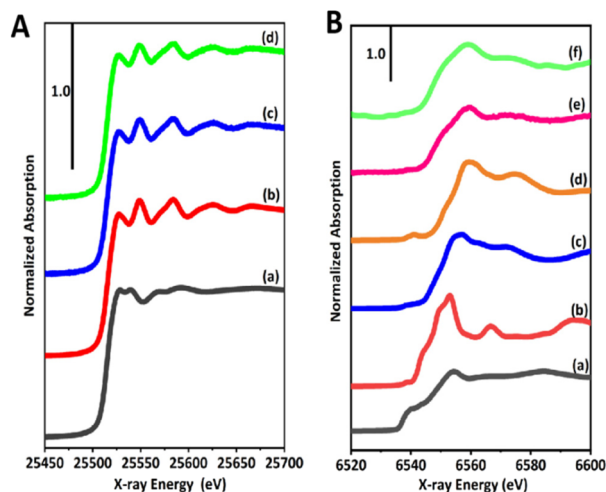
**Fig. 5** Formation rates of H<sub>2</sub> (red), O<sub>2</sub> (blue), CO (green), and H<sub>2</sub>O<sub>2</sub> (violet), selectivity for CO evolution ( $S_{CO}$ , square), and  $R(e^-/h^+)$  (black dot) in the photocatalytic reaction tests for CO<sub>2</sub> reduction with water as an electron donor over various BT4(20,1273) samples.

### 3.3. Dual cocatalyst for BaTi<sub>4</sub>O<sub>9</sub>

Photocatalytic reaction tests were performed over the BT4 samples with different cocatalysts Ag, MnO<sub>*x*</sub>, and Ag-MnO<sub>*x*</sub>, as shown in Fig. 5. Bare BT4 was confirmed as a satisfactory photocatalyst for water splitting, with stoichiometric evolution of H<sub>2</sub> and O<sub>2</sub>.<sup>23</sup> As expected, it was confirmed that the Ag/BT4(20,1273) sample produced CO and O<sub>2</sub> with a high formation rate and high CO selectivity of 94%.<sup>28</sup> It was found that the MnO<sub>*x*</sub>/BT4(20,1273) sample exhibited a similar H<sub>2</sub> formation rate, but less O<sub>2</sub> formation compared with bare BT4(20,1273). Thus, the resulting solution was filtered after the photocatalytic reaction, and was then used for the detection of H<sub>2</sub>O<sub>2</sub> formation by potassium permanganate titration, which is a conventional method for testing H<sub>2</sub>O<sub>2</sub> formation.<sup>33</sup>

From the results shown in Fig. S8,† H<sub>2</sub>O<sub>2</sub> was proved as a possible oxidative product. This confirmed that the MnO<sub>*x*</sub> cocatalyst contributed to H<sub>2</sub>O<sub>2</sub> formation (eqn (4)), and indicated that the MnO<sub>*x*</sub> cocatalyst on BT4 can function as a cocatalyst for water oxidation with holes. The formation of hydrogen peroxide has also been reported for Ag-MnO<sub>*x*</sub>/CaTiO<sub>3</sub><sup>33</sup> and Ag-CoO<sub>*x*</sub>/CaTiO<sub>3</sub><sup>46</sup> in photocatalytic CO<sub>2</sub> reduction with water in aqueous solution of NaHCO<sub>3</sub>. The CO formation rate for the Ag-MnO<sub>*x*</sub>/BT4(20,1273) sample was 2 times higher than that of the Ag/BT4(20,1273) sample. The O<sub>2</sub> production rate was less than half of the CO formation rate, and production of H<sub>2</sub>O<sub>2</sub> was found. In the calculation with formed H<sub>2</sub>O<sub>2</sub>, the  $R(e^-/h^+)$  value was nearly unity (Fig. 5). In a long-time reaction test with the Ag-MnO<sub>*x*</sub>/BT4(20,1273) sample (Fig. S9)†, after the initial period of 3.5 h, the formation rates for the products became constant, which indicated the stability of the photocatalyst.



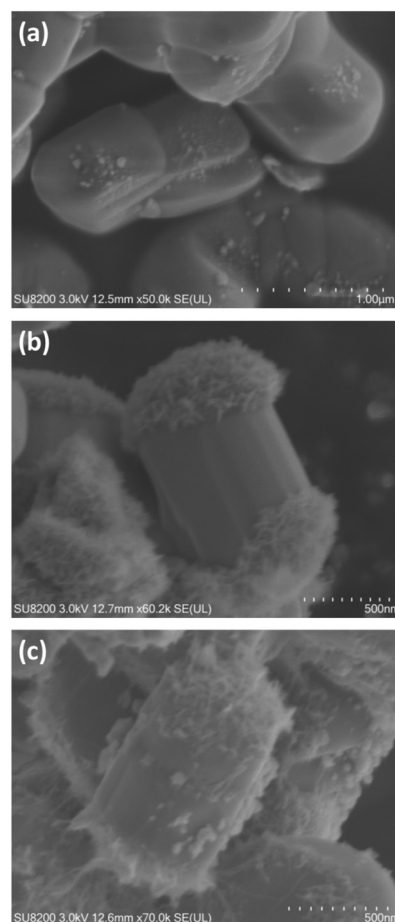


**Fig. 6** [A] Ag K-edge XANES of (a) Ag<sub>2</sub>O, (b) Ag foil, (c) the Ag(1)/BT4 sample, and (d) the Ag(1)-MnO<sub>x</sub>(0.492)/BT4 sample. [B] Mn K-edge XANES of (a) Mn foil, (b) MnO, (c) Mn<sub>2</sub>O<sub>3</sub>, (d) MnO<sub>2</sub>, (e) the MnO<sub>x</sub>(0.580)/BT4 sample, and (f) the Ag(1)-MnO<sub>x</sub>(0.492)/BT4 sample. The BT4 sample used was the BT4(20,1273) sample.

The chemical states of the dual cocatalyst of Ag and Mn species were investigated by X-ray adsorption spectroscopy (XAS). Fig. 6A shows the Ag K-edge X-ray absorption near edge structure (XANES) of the prepared samples and corresponding references, Ag foil, and Ag<sub>2</sub>O. The shape and intensity of XANES with the first peak at 25 526 eV for the samples were in good agreement with that of Ag foil, but were clearly distinguished from that of Ag<sub>2</sub>O (Fig. 6A and S10<sup>†</sup>), revealing that the Ag cocatalyst existed in the form of metallic Ag in the Ag(1)/BT4(20,1273) and Ag(1)-MnO<sub>x</sub>(0.492)/BT4(20,1273) samples. This suggests that the simultaneous deposition of Mn species would not influence the metallic state of the Ag cocatalyst.

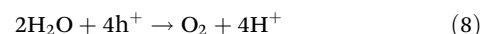
As for the Mn species deposited on the prepared samples (Fig. 6Be and f), the spectra shapes were not strictly consistent with Mn foil or any state of typical manganese oxides including MnO, Mn<sub>2</sub>O<sub>3</sub>, and MnO<sub>2</sub> (Fig. 6Ba–d). In detail, the main peak at 6559 eV was similar to that for MnO<sub>2</sub> (Fig. 6Bd), while the edge at 6543 eV was likely that for Mn<sub>2</sub>O<sub>3</sub> (Fig. 6Bc), meaning that the Mn<sup>2+</sup> precursor was likely oxidized to a mixture of Mn<sup>3+</sup> and Mn<sup>4+</sup>. It was noted that the XANES of MnO<sub>x</sub>(0.580)/BT4(20,1273) and Ag(1)-MnO<sub>x</sub>(0.492)/BT4(20,1273) were similar to each other, indicating that the addition of silver will not greatly affect the chemical state of the MnO<sub>x</sub> species. Thus, the MnO<sub>x</sub> species deposited on the surface of BT4 by the photodeposition (PD and SPD) method exhibited a unique feature and local structure that was different from those of typical manganese oxides used as references regardless of the photodeposition of Ag species.

The location of photodeposited metal and metal oxide species assist in understanding the function of the facets of the photocatalyst crystals, as mentioned above.<sup>47</sup> The morphology of the BT4(20,1273) crystal was typically that of rectangular or cylindrical crystals (Fig. 7, S7 and S11<sup>†</sup>), and it was



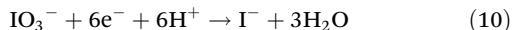
**Fig. 7** SEM images of the samples: (a) Ag(1)/BT4, (b) MnO<sub>x</sub>(2.4)/BT4, and (c) Ag(1.499)-MnO<sub>x</sub>(0.523)/BT4. The BT4 sample used was BT4(20,1273).

not well controlled at present. When the Ag cocatalyst was photodeposited on the cylindrical BT4(20,1273) crystal (Fig. 7a), Ag NPs were formed on the sides of the cylindrical BT4 crystals. Namely, photogenerated electrons migrated to these facets, and the reduction of Ag<sup>+</sup> cations preferentially took place on these facets to form metallic Ag nanoparticles, while the photogenerated holes oxidized water, as described by eqn (7) and (8):



When the MnO<sub>x</sub> cocatalyst was photodeposited with sodium iodoxide (NaIO<sub>3</sub>) as a sacrificial reagent (electron acceptor), petal-like MnO<sub>x</sub> species were selectively formed on the top and bottom of the cylindrical BT4 crystal (Fig. 7b), and the places and morphology were clearly different from those of photodeposited Ag NPs (Fig. 7a). The MnO<sub>x</sub> cocatalyst was deposited through oxidation of Mn(II) cations with positive holes on the oxidative facets (eqn (9) and (10)), where MnO<sub>2</sub> is representatively used in the equation instead of MnO<sub>x</sub>:





When these Ag and MnO<sub>x</sub> cocatalysts were deposited as a dual cocatalyst using the SPD method (Fig. 7c), the side and top/bottom were selectively decorated with Ag NPs and MnO<sub>x</sub> species, respectively, resulting in full coverage of the cylindrical BT4 crystal. Thus, we can recognize the position of the reductive and oxidative facets according to the position of the Ag NPs and MnO<sub>x</sub> species, as shown in Fig. 7. This indicates that the photoexcited electrons can preferentially migrate to the Ag NPs, and the photoformed holes can dominantly move to the MnO<sub>x</sub> species. The photocatalytic activity was actually enhanced by the selectively photodeposited Ag and MnO<sub>x</sub> cocatalysts (Fig. 5). Therefore, we can draw the conclusion that the charge transfer to certain facets promotes the reaction, as shown in Fig. 8.

These observations revealed that the side and top/bottom of the cylindrical BT4 crystal are the reductive and oxidative facets, respectively. This fact was further confirmed on the rectangular-shaped BT4(20,1273) crystals by photodeposition of Pt<sup>4+</sup> and Pb<sup>2+</sup>, as shown in Fig. S11.†

It was reported that although the interaction between Ag and Mn species will disturb CO<sub>2</sub> photoreduction with H<sub>2</sub>O, the facet-selective photodeposition of Ag and Mn species prevented the interaction between Ag and Mn species, which led to the high photocatalytic performance.<sup>46</sup> Because the photocatalytic activity was enhanced (Fig. 5), it could be considered that the Ag nanoparticles on the reductive facets and the MnO<sub>x</sub> on the oxidative facets play important roles in the photocatalytic performance. The selective photodeposition of the Ag-MnO<sub>x</sub> dual cocatalyst on the present BT4 crystals greatly improved the photocatalytic activity, and the property of the MnO<sub>x</sub> species changed the selectivity for the oxidation with holes to form H<sub>2</sub>O<sub>2</sub>.

### 3.4. Proposed mechanism

As mentioned above, in the photodeposition of the cocatalysts on the BT4 crystal, Ag<sup>+</sup> cations are reduced by the photoexcited electrons and then deposited as Ag NPs on the reductive facet of the BT4 crystal, while Mn<sup>2+</sup> cations are oxidized by the holes

to form petal-like MnO<sub>x</sub> species on the oxidative facets. Thus, in the photocatalytic reaction with the Ag-MnO<sub>x</sub>/BT4 photocatalyst, the photoexcited electrons will reach the Ag NPs, and holes will go to the MnO<sub>x</sub> species to promote the reductive and oxidative reactions, respectively. Ag selectively promotes CO<sub>2</sub> photoreduction to CO (eqn (1)), while MnO<sub>x</sub> promotes the oxidation of water to O<sub>2</sub> or H<sub>2</sub>O<sub>2</sub> (eqn (3) and (4)). Thus, the Ag and MnO<sub>x</sub> species can individually participate in CO<sub>2</sub> photoreduction and water oxidation, as illustrated in Fig. 8. The efficient consumption of electrons and holes reduces recombination, resulting in high photocatalytic performance.

## 4. Conclusions

Several barium titanate compounds with different compositions were prepared for examination. BaTi<sub>n</sub>O<sub>2n+1</sub> (*n* = 1, 4) and Ba<sub>2(n-1)</sub>Ti<sub>4n+1</sub>O<sub>10n</sub> (*n* = 2, 3, 4) were prepared by a solid-state reaction method, and BaTi<sub>2</sub>O<sub>5</sub> was prepared by a flux method, while pure BaTi<sub>5</sub>O<sub>11</sub> was not obtained under the present conditions. Among the prepared barium titanates, it was demonstrated that BaTi<sub>4</sub>O<sub>9</sub> (BT4) was a satisfactory photocatalyst for CO<sub>2</sub> reduction with water. The formation of clear reductive and oxidative facets on the cylindrical (or rectangular) BT4 crystals was realized by optimization of preparation conditions for BT4.

When an Ag-MnO<sub>x</sub> dual cocatalyst is loaded by the photodeposition method, Ag NPs and petal-like MnO<sub>x</sub> species are separately loaded on the reductive and oxidative facets, respectively. Such prepared Ag-MnO<sub>x</sub>/BT4 crystal photocatalysts can exhibit excellent photocatalytic activity for CO<sub>2</sub> reduction with water, where CO is formed as a reduction product, and O<sub>2</sub> and H<sub>2</sub>O<sub>2</sub> are formed as oxidation products.

This work demonstrates the potential of BaTi<sub>4</sub>O<sub>9</sub> crystal to act as a photocatalyst, and the efficient property of the Ag-MnO<sub>x</sub> dual cocatalyst for the BT4 photocatalyst, which will contribute to the further development of photocatalysts for CO<sub>2</sub> reduction with water to obtain CO as a valuable intermediate for industrial chemistry by utilization of solar energy.

## Author contributions

Shuwei Liu: conceptualization, investigation, writing – original draft. Hongxuan Qiu: conceptualization, investigation, methodology. Akira Yamamoto: investigation, methodology, funding acquisition. Hisao Yoshida: conceptualization, funding acquisition, project administration, supervision, writing – review and editing.

## Conflicts of interest

There are no conflicts to declare.

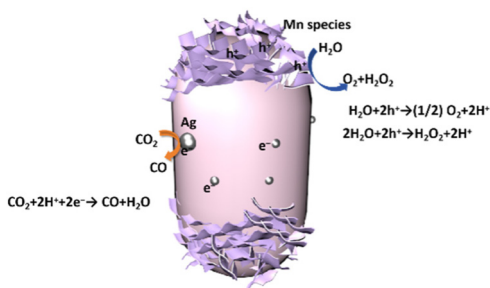


Fig. 8 Schematic drawing of photocatalytic CO<sub>2</sub> reduction with water as an electron donor on the Ag-MnO<sub>x</sub>/BT4 crystal photocatalyst.



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## References

- G. A. Florides and P. Christodoulides, *Environ. Int.*, 2009, **35**, 390–401.
- J. Zhao, X. Wang, Z. Xu and J. S. C. Loo, *J. Mater. Chem. A*, 2014, **2**, 15228–15233.
- H. Huang, J. Zhao, B. Weng, F. Lai, M. Zhang, J. Hofkens, M. B. J. Roeffaers, J. A. Steele and J. Long, *Angew. Chem., Int. Ed.*, 2022, **61**, e202204563.
- S. Wang, X. Han, Y. Zhang, N. Tian, T. Ma and H. Huang, *Small Struct.*, 2020, **2**, 2000061.
- S. Abello and D. Montane, *ChemSusChem*, 2011, **4**, 1538–1556.
- D. K. Chauhan, N. Sharma and K. Kailasam, *Mater. Adv.*, 2022, **3**, 5274–5298.
- R. Pang, K. Teramura, H. Tatsumi, H. Asakura, S. Hosokawa and T. Tanaka, *Chem. Commun.*, 2018, **54**, 1053–1056.
- K. Iizuka, T. Wato, Y. Miseki, K. Saito and A. Kudo, *J. Am. Chem. Soc.*, 2011, **133**, 20863–20868.
- Z. Wang, K. Teramura, S. Hosokawa and T. Tanaka, *Appl. Catal., B*, 2015, **163**, 241–247.
- Z. Huang, K. Teramura, S. Hosokawa and T. Tanaka, *Appl. Catal., B*, 2016, **199**, 272–281.
- T. Takayama, A. Iwase and A. Kudo, *Bull. Chem. Soc. Jpn.*, 2015, **88**, 538–543.
- Z. Wang, K. Teramura, S. Hosokawa and T. Tanaka, *J. Mater. Chem. A*, 2015, **3**, 11313–11319.
- M. Yamamoto, T. Yoshida, N. Yamamoto, T. Nomoto, Y. Yamamoto, S. Yagi and H. Yoshida, *J. Mater. Chem. A*, 2015, **3**, 16810–16816.
- S. Iguchi, K. Teramura, S. Hosokawa and T. Tanaka, *Catal. Sci. Technol.*, 2016, **6**, 4978–4985.
- A. Anzai, N. Fukuo, A. Yamamoto and H. Yoshida, *Catal. Commun.*, 2017, **100**, 134–138.
- X. Zhu, A. Yamamoto and H. Yoshida, *Dalton Trans.*, 2021, **50**, 7976–7983.
- G. Panthi and M. Park, *J. Energy Chem.*, 2022, **73**, 160–188.
- X. Zhang, X. Wang, J. Chai, S. Xue, R. Wang, L. Jiang, J. Wang, Z. Zhang and D. D. Dionysiou, *Appl. Catal., B*, 2020, **272**, 119017.
- Y. Cui, J. Briscoe, Y. Wang, N. V. Tarakina and S. Dunn, *ACS Appl. Mater. Interfaces*, 2017, **9**, 24518–24526.
- H. Fan, H. Li, B. Liu, Y. Lu, T. Xie and D. Wang, *ACS Appl. Mater. Interfaces*, 2012, **4**, 4853–4857.
- Z. Zhao, D. Wang, R. Gao, G. Wen, M. Feng, G. Song, J. Zhu, D. Luo, H. Tan, X. Ge, W. Zhang, Y. Zhang, L. Zheng, H. Li and Z. Chen, *Angew. Chem., Int. Ed.*, 2021, **60**, 11910–11918.
- Y. Yamashita, M. Tada, M. Kakihana, M. Osada and K. Yoshida, *J. Mater. Chem.*, 2002, **12**, 1782–1786.
- Y. Hiramachi, H. Fujimori, A. Yamakata and Y. Sakata, *ChemCatChem*, 2019, **11**, 6213–6217.
- M. Kohno, T. Kaneko, S. Ogura, K. Sato and Y. Inoue, *J. Chem. Soc., Faraday Trans.*, 1998, **94**, 89–94.
- G. Yang, J. Xiong, M. Lu, W. Wang, W. Li, Z. Wen, S. Li, W. Li, R. Chen and G. Cheng, *J. Colloid Interface Sci.*, 2022, **624**, 348–361.
- M. Kohno, S. Ogura and Y. Inoue, *J. Mater. Chem.*, 1996, **6**, 1921–1924.
- Y. Inoue, T. Niiyama, Y. Asai and K. Sato, *J. Chem. Soc., Chem. Commun.*, 1992, 579–580.
- A. Anzai, A. Yamamoto and H. Yoshida, *Catal. Lett.*, 2021, **152**, 2498–2506.
- T. Ohno, K. Sarukawa and M. Matsumura, *New J. Chem.*, 2002, **26**, 1167–1170.
- R. Li, F. Zhang, D. Wang, J. Yang, M. Li, J. Zhu, X. Zhou, H. Han and C. Li, *Nat. Commun.*, 2013, **4**, 1432.
- J. Zhu, F. Fan, R. Chen, H. An, Z. Feng and C. Li, *Angew. Chem., Int. Ed.*, 2015, **54**, 9111–9114.
- X. Zhu, A. Yamamoto, S. Imai, A. Tanaka, H. Kominami and H. Yoshida, *Chem. Commun.*, 2019, **55**, 13514–13517.
- T. Soltani, A. Yamamoto, S. P. Singh, A. Anzai, E. Fudo, A. Tanaka, H. Kominami and H. Yoshida, *ACS Appl. Energy Mater.*, 2021, **4**, 6500–6510.
- E. Song, D. H. Kim, E. J. Jeong, M. Choi, Y. Kim, H. J. Jung and M. Y. Choi, *Environ. Res.*, 2021, **202**, 111668.
- S. Zhu, Y. Dai and X. Pei, *J. Mater. Sci.: Mater. Electron.*, 2022, **33**, 16406–16413.
- J. Fu, Y. Hou, M. Zheng and M. Zhu, *CrystEngComm*, 2017, **19**, 1115–1122.
- H. Zhou, H. Wang, Y. Chen, K. Li and X. Yao, *J. Am. Ceram. Soc.*, 2008, **91**, 3444–3447.
- Y. Miseki and A. Kudo, *ChemSusChem*, 2010, **4**, 245–251.
- J. Tauc, R. Grigorovici and A. Vancu, *Phys. Status Solidi*, 1966, **15**, 627–637.
- D. E. Scaife, *Sol. Energy*, 1980, **25**, 41–54.
- X. Lin, J. Wu, X. Lu, Z. Shan, W. Wang and F. Huang, *Phys. Chem. Chem. Phys.*, 2009, **11**, 10047–10052.
- D. E. Rase and R. Roy, *J. Am. Ceram. Soc.*, 2006, **38**, 102–113.
- T. Ishii, R. Takioka, H. Yasumura, A. Yamamoto and H. Yoshida, *Catal. Today*, 2024, **429**, 114476.
- H. Yoshida, L. Zhang, M. Sato, T. Morikawa, T. Kajino, T. Sekito, S. Matsumoto and H. Hirata, *Catal. Today*, 2015, **251**, 132–139.
- H. Yoshida, M. Sato, N. Fukuo, L. Zhang, T. Yoshida, Y. Yamamoto, T. Morikawa, T. Kajino, M. Sakano, T. Sekito, S. Matsumoto and H. Hirata, *Catal. Today*, 2018, **303**, 296–304.
- H. Qiu, A. Yamamoto, E. Fudo, A. Tanaka, H. Kominami and H. Yoshida, *ACS Appl. Energy Mater.*, 2023, **6**, 11592–11601.
- X. Zhu, A. Yamamoto, S. Imai, A. Tanaka, H. Kominami and H. Yoshida, *Appl. Catal., B*, 2020, **274**, 119085.

