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# Biomimetic antimicrobial polymers: recent advances in molecular design

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The increasing prevalence of antibiotic-resistant bacterial infections, coupled with the decline in the number of new antibiotic drug approvals, has created a therapeutic gap that portends an emergent public health crisis. Since the 1980s, host defense peptides (HDPs) have been recognized as antibacterial compounds that do not induce resistance, but are hampered by their high cost and lack of synthetic scalability. Starting in the early 2000s, synthetic (co)polymers have been designed to mimic the salient physicochemical features of HDPs. These polymers have shown broad-spectrum antimicrobial activity, rapid bactericidal kinetics, and a very low propensity to induce resistance. Systematic optimization of the (co)polymer composition, chain length, hydrophobicity, and cationic charge has generated select examples that are also highly biocompatible (non-hemolytic and non-cytotoxic *in vitro*). These polymers are derived from inexpensive feedstocks and are produced using cost-effective, scalable processes. Accordingly, such polymers may be viewed as early stage pre-clinical candidates for potential use in pharmaceutical or therapeutic applications. In this review, we focus on the key macromolecular design principles that have been gleaned from more than a decade of structure–activity relationship (SAR) studies, as well as some key mechanistic investigations, across this multidisciplinary field. A fundamental understanding of these functional (co)polymers has arisen from a convergence of ideas in polymer chemistry, microbiology, and biophysics. In this context, we emphasize the recent advances from the past few years and emerging opportunities surrounding the rapidly growing field of HDP-mimetic antimicrobial polymers.

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## 1. Motivation and background

### 1.1. Host defense peptides

The rise of antibiotic drug resistance in infectious pathogens presents one of the most daunting challenges facing modern medicine.<sup>1</sup> Tens of millions of lives are saved annually by the routine use of broad-spectrum antibiotics. Without these indispensable drugs, society would abruptly return to the *status quo ante* of the 1930s, an unimaginably bleak prospect by modern standards. For example, procedures such as caesarean section birth carried a disturbingly high risk of mortality prior to the advent of antibiotics.

Host defense peptides (HDPs) are components of innate immunity expressed by all multicellular organisms<sup>2–4</sup> (representative structural examples are shown in Fig. 1). The putative function of an HDP, at least in part, is to kill bacteria without harming host cells and without inducing resistance. The ability to exert broad-spectrum activity without causing resistance is related to their mechanism of action, which is



**Fig. 1** Molecular models of four HDPs. The hydrophobic residues are red and the hydrophilic residues are blue. There is a lack of conserved secondary structure but hydrophobic residues appear to cluster into distinct domains. The images were generated using the RCSB Protein Databank (<http://www.rcsb.org/pdb/home/home.do>).

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thought to involve membrane permeabilization.<sup>5–7</sup> Instead of a precise “lock and key” mechanism, the membrane disruption pathway is much less specific and thus more challenging for bacteria to circumvent. Cationic residues (Lys and Arg) in the HDPs experience electrostatic attraction to anionic components of the bacterial cell membrane whereas their abundant hydrophobic residues readily insert in the non-polar membrane core. Such peptides have been shown to permeabilize model liposome vesicles<sup>8</sup> and to induce leakage of potassium ions and enzymes from bacterial cells, presumably leading to concomitant cell death. Electron microscopy,<sup>9</sup> confocal microscopy, and even high-speed AFM<sup>10</sup> of bacterial cells upon exposure to HDPs and related compounds also support the notion of membrane permeabilization. It is important to note that the antibacterial activity of HDPs is in fact more specific than originally thought; there is convincing evidence that defensins, for example, sequester microbe-specific lipid receptors (lipid II), inhibiting cell wall biosynthesis.<sup>11</sup> In this review, however, we focus on the HDP-mimetic design and optimization of synthetic polymers that aim to capture the essence of a non-specific membrane-disruption mechanism.

One of the earliest discoveries of an HDP was Magainin-2, isolated from the skin of the African clawed frog *Xenopus laevis*, by M. Zasloff in 1987.<sup>12</sup> Since that time, thousands of other antimicrobial peptides have been discovered and it is now understood that they exist in all multicellular organisms. Wang and Wang maintain a detailed online database of all known antimicrobial peptides (AMPs), with 2903 current entries.<sup>13–15</sup> Among this class, there is a remarkable lack of conserved sequence or preferred secondary structure. Rather, the commonalities seem to be *physiochemical* in nature: AMPs (1) possess cationic charge at neutral pH, (2) contain a substantial fraction of hydrophobic residues, and (3) are relatively short chain peptides (Fig. 2). The hydrophobic groups are key to their interaction with phospholipid bilayers and the cationic charge is considered crucial for their electrostatic attraction to anionic components of the bacterial cell surface. Non-natural analogs of HDPs, such as all- $\alpha$  peptides,<sup>16,17</sup>  $\beta$ -peptides,<sup>18–20</sup> and peptoids,<sup>21,22</sup> successfully reproduced the antibacterial activity and low toxicity to human cells that are the hallmarks of HDP

efficacy, simply by designing sequences that possess the requisite cationic amphiphilicity. These abiotic compounds are referred to as antimicrobial peptides “AMPs”, but they are not HDPs.

As an interesting side note, a small fraction of AMPs appear to bear a net *negative* charge at neutral pH. One such example is the surfactant-associated anionic peptide (SAAP, charge  $-7$ ) from *Ovis aries*, which only exerts activity when complexed with  $Zn^{2+}$  ions.<sup>23</sup> Another anionic example is microplusin (charge  $-8$ ), which requires binding to  $Cu^{2+}$  as a cofactor to exert activity against bacteria.<sup>24</sup> Although the number of such anionic AMPs represents less than 3% of the total, it is possible that they possess some unique characteristics worthy of biomimicry.

The conspicuous lack of any conserved sequence or defined secondary structure led DeGrado's group to the hypothesis that synthetic (co)polymers – with all their structural imperfections and heterogeneities – could perhaps be designed to capture the essential physiochemical features of HDPs that are key to their antimicrobial activity. Indeed, this strategy has been widely employed by numerous research groups over the past decade with a great deal of promise. In this review, we will focus on the key structural determinants of antibacterial activity, and toxicity to human cells, in this class of peptide-mimetic macromolecules. We begin with summarizing the classical design approaches to cationic amphiphilicity and then expand on this idea to include the most recent advances. We also summarize the relatively limited, but rapidly growing, number of studies on *in vivo* activity, and finally we review the current understanding of their mechanism(s) of bactericidal action. This article is by no means intended to represent an exhaustive summary of all the valuable contributions to the field, and the keen reader is encouraged to reference a large number of other excellent reviews.<sup>25–35</sup> Here, we place substantial emphasis on new work published in the past two years, which are not covered by numerous prior reviews, in addition to a summary of the major achievements of the past 10 years that have laid the foundation for future development.

## 1.2. Convergent trends: HDPs and benzalkonium chlorides

Concurrently with the discovery of HDPs, conceptually and structurally related polymeric disinfectants were being devel-



Fig. 2 Despite their lack of conserved sequence or secondary structure, there are three common physiochemical characteristics of antimicrobial peptides: (a) hydrophobicity, (b) cationic charge, and (c) short chain length. The data were collected from The Antimicrobial Peptide Database (<http://aps.unmc.edu/AP/structure.php>).



oped in the 1980s,<sup>36</sup> although there appears to have been little connection made between these two areas initially. Common commercial biocides containing the so-called “benzalkonium chlorides” (BACs) are cationic surfactants composed of a benzylic quaternary ammonium salt (QAS) and a long alkyl chain (C<sub>8</sub>H<sub>17</sub> to C<sub>18</sub>H<sub>37</sub>). The monomeric surfactants indiscriminately lyse biomembranes at high concentrations, regardless of cell type (*i.e.* they are “biocidal”). Thus, BACs are bactericidal but also toxic to mammals.

In 1984, Ikeda and co-workers reported the polymerization of a styrene derivative that contained a pendant BAC surfactant in the side chain (Fig. 3).<sup>37</sup> Not surprisingly, both the styrenic monomer and the polymer thereof exerted antibacterial activity, albeit in the mg mL<sup>-1</sup> concentration range. The toxicity data were not shown, but the intensely hydrophobic nature of the polymer side chains, as well as the high concentrations of the polymer employed, suggests that these materials would almost certainly exert surfactant-like membrane lysis regardless of the cell type.

During the late 1990s, HDPs inspired various peptidomimetics such as  $\beta$ -peptides,<sup>20,38</sup>  $\alpha/\beta$ -peptides<sup>39</sup> and peptoids<sup>40</sup> with antimicrobial activity. In parallel, poly(BAC)s inspired biocidal polymers with a variety of cationic and hydrophobic substituents. By the mid-2000s, the two fields effectively converged

into *HDP-mimetic polymers*, and the field began to experience rapid growth that continues today (Fig. 4). The convergence of trends is a prime example of how multidisciplinary science can vertically advance technologies at the interface of conventionally unrelated fields.

In this comparative context, it is instructive to consider the bee venom toxin melittin.<sup>41</sup> This peptide is 26 a.a. residues in length, contains 46% hydrophobic residues, and bears a +6 cationic charge at neutral pH. Viewing these data alone, the structure may appear to have more in common with HDPs than with biocidal polymers. However, the activity profile of melittin is more similar to that of a QAS-containing disinfectant: it completely lyses both bacterial cells and human red blood cells alike at approximately the same concentrations. Such a comparison invites one to ponder: What are the key design principles that distinguish a cell-selective antibacterial macromolecule from a biocidal one? Can biocidal polymers be re-optimized to confer cell-type selectivity?

### 1.3. Terminology

Before a discussion of structure–activity relationships (SAR), the technical jargon of this field deserves an explanation for the non-expert (Table 1). The MIC, a widely used metric, is the lowest concentration of a polymer that prevents the prolifer-



**Fig. 3** The evolution of antimicrobial polymers. Molecular design principles were gleaned from a convergence of ideas on antimicrobial peptides and synthetic polymer disinfectants. HDP-inspired polymers are intended to combine the advantages of peptide activity with the cost-effectiveness and scalability of synthetic polymer chemistry, without the need for precise sequence, unimolecular chain length, or defined secondary structure.





**Fig. 4** Number of publications as a function of time that contain the phrase “antimicrobial polymer” via ISI Web of Science. These data include the HDP-mimetic polymer literature (the scope of this review), as well as biocidal polymers, biopolymers, polymer/drug conjugates, and composites of polymers with antibacterial inorganics.

ation of bacteria. The MIC is the *bacteriostatic* concentration at which bacteria do not *grow* (but may or may not be *dead*). In contrast, the MBC reflects the *bactericidal* concentration at which most or all of the bacterial cells are confirmed dead. The MBC represents at least a 99.9% (or 99.99%) reduction in colony-forming units (CFU) per mL. The MBC test is more labor intensive because it requires colony counting on nutrient agar plates, but is more informative. Frequently for antibacterial polymers, the MIC and MBC values are identical or very similar, but it must always be the case that  $MIC \geq MBC$  for a given set of test conditions (it is not possible to kill a cell but also fail to inhibit its growth, whereas the converse is indeed possible).

The most commonly used measure of toxicity to human cells is the hemolysis assay. At the  $HC_{50}$  concentration, a compound induces 50% release of hemoglobin from a suspension of RBCs. This colorimetric assay is amenable to high throughput, and provides a quantification of the membrane-lytic activity against mammalian cell membranes. Cytotoxicity

against mammalian cell cultures *in vitro* is less frequently reported but arguably more informative. The concentration of the polymer that causes a 50% reduction in cell viability is termed the  $LC_{50}$ . Various cell lines (HeLa, HEP-2, Cos7, Jurkat, *etc.*) along with various high-throughput colorimetric assay kits (LDH, MTT, XTT, *etc.*) are routinely employed to that end.

It is important to recognize that all the benchmark activity values are highly assay-dependent.<sup>42</sup> The same polymer can show different MIC values, for example, based on the type of assay media, the initial concentration of bacteria in the inoculum (typically  $\sim 10^6$  CFU mL<sup>-1</sup>), and – *surprisingly* – even the material of which the microplate is made. Melittin gives a 4× higher MIC value when tested on “tissue-culture treated” polystyrene microplates compared to polypropylene microplates, all else being equal. Similarly, the  $HC_{50}$  value can vary significantly depending on the pH and the concentration of the RBCs in the buffer (typically  $10^7$  cells per mL, with higher RBC concentrations resulting in higher  $HC_{50}$ ). For these reasons, it is important to view MIC and  $HC_{50}$  literature values *relative to a standard of known activity* (such as Magainin-2 or Melittin) tested under strictly identical assay conditions.

The terms “antibacterial” and “antimicrobial” are distinct: the former is a compound that inhibits (or kills) bacterial cells, whereas the latter is one that inhibits (or kills) all microorganisms, including bacteria, yeast and viruses. Although sometimes used interchangeably, “antibacterial” is the more accurate term for a polymer that has not been tested against yeast or viruses, strictly speaking. The terms “host defense peptide” (HDP) and “antimicrobial peptide” (AMP) are also subtly distinct: the former refers to the naturally-occurring components of innate immunity in host organisms, whereas the latter is any peptide with antimicrobial activity, whether it be a natural HDP or a synthetic peptide. Finally, the term “biocidal” refers to a compound that indiscriminately kills both prokaryotic and eukaryotic cells.

In this article, we denote the concentration range of 1–10  $\mu\text{g mL}^{-1}$  as “very potent” activity, whereas the range of 10–100  $\mu\text{g mL}^{-1}$  is called “moderate” activity, 100–1000  $\mu\text{g mL}^{-1}$  is “weak” and  $>1000 \mu\text{g mL}^{-1}$  is “inactive”. These ranges are only intended as a rough guide to facilitate our discussion and are by no means intended as strict rules.

**Table 1** Technical jargon widely used in the antimicrobial polymer field

Term	Definition	Comments
Minimum inhibitory concentration (MIC)	The lowest concentration of a polymer that completely inhibits the <i>growth</i> of a microorganism in nutrient media.	<ul style="list-style-type: none"> <li>• Turbidity-based, high-throughput assay</li> <li>• Lower MIC = <i>better</i> activity</li> <li>• <i>Bacteriostatic</i> activity</li> </ul>
Minimum bactericidal concentration (MBC)	The lowest concentration of a polymer that reduces bacterial cell viability by at least 99.9% of the initial inoculum.	<ul style="list-style-type: none"> <li>• Colony counting, low-throughput assay</li> <li>• Lower MBC = <i>better</i> activity</li> <li>• <i>Bactericidal</i> activity</li> </ul>
Hemolytic concentration ( $HC_{50}$ )	The concentration that induces 50% release of hemoglobin from red blood cells in buffer.	<ul style="list-style-type: none"> <li>• Colorimetric, high-throughput assay</li> <li>• Lower <math>HC_{50}</math> = <i>worse</i> toxicity</li> </ul>
Lethal concentration ( $LC_{50}$ )	The concentration that reduces mammalian cell viability by 50% in culture media.	<ul style="list-style-type: none"> <li>• Colorimetric, high-throughput assay</li> <li>• Lower <math>LC_{50}</math> = <i>worse</i> toxicity</li> </ul>



## 2. Structure–activity relationships (SAR)

### 2.1. Fundamental design principles

**Amphiphilic balance.** Of all the SAR principles discussed in this field, the most universally embraced idea is the need for some “balance” between the cationic and hydrophobic components in the polymer structure. Excessive hydrophobicity causes high toxicity and poor solubility, whereas polycations with very low hydrophobicity are typically not potent antibacterials and also tend to induce aggregation of RBCs (hemagglutination). Between these two extremes, there are certain formulations that provide the desired balance of hydrophobic and cationic character. For example, among the poly(methacrylate) random copolymers of Kuroda and co-workers,<sup>43</sup> a ~10-mer with 40% methyl side chains and 60% aminoethyl side chains will exhibit the desired combination of potent antibac-

terial activity and minimal hemolytic toxicity, whereas the cationic homopolymer (containing 0% methyl side chains) is inactive against *E. coli*, and the highly hydrophobic copolymers (with more than ~70% methyl side chains) are highly hemolytic and poorly soluble (Fig. 5). Qualitatively similar results were obtained by the optimization of the hydrophobic and cationic composition in poly(norbornene)s,<sup>44</sup> nylon-3 copolymers,<sup>45,46</sup> and others, although each unique platform requires independent optimization to find the right balance.

It should be acknowledged that the concept of “amphiphilic balance” in fact predates HDP-mimetic polymers; it was extensively discussed for biocidal polymers since the 1990s.<sup>36</sup> Benzalkonium chlorides, and their polymeric derivatives, contain very long alkyl chains (up to C<sub>18</sub>) attached to their cationic QAS groups and are toxic to human cells. In stark contrast, the *most* hydrophobic amino acid is isoleucine (with a C<sub>4</sub> side chain). Thus, the use of modestly hydrophobic side chain groups (C<sub>1</sub>–C<sub>4</sub>) in HDP-mimetic polymers is a crucial factor determining their cell-type selectivity. It has been shown that the overall hydrophobicity of the polymer will dictate the hemolytic toxicity.<sup>43</sup> The fundamental design principle is therefore quite simple: incorporate the *minimum* amount of hydrophobic content needed to confer antibacterial activity. Fig. 6 shows the diversity of structural modifications that have been carried out on the polymethacrylate platform in order to optimize the balance of hydrophobic and cationic character.

**Cationic group structure.** Inspired by lysine-rich HDPs, many antibacterial synthetic polymers contain pendant primary amine groups as their source of cationic charge, in contrast to the quaternary ammonium salt (QAS) groups that are ubiquitous in biocidal polymers inspired by BACs. A direct comparison of primary *versus* quaternary ammonium groups as the cationic moieties in polymethacrylates (with the same  $M_n$ ,  $D$  and % copolymer composition for direct comparison) revealed that the polymers bearing primary ammonium groups outperformed their tertiary and quaternary analogues in terms of more potent antibacterial activity and lower toxicity



**Fig. 5** Amphiphilic balance is a widely employed concept for the optimization of copolymer composition in antibacterial polymers. In random copolymers of methacrylates, the ratio of cationic to hydrophobic comonomer dictates the antibacterial and hemolytic activities.



**Fig. 6** The broad diversity of cationic and hydrophobic group structures, and their relative spatial arrangements, have been extensively tailored in the poly(methacrylate) design platform towards the optimization of biological activity.



to human cells.<sup>47</sup> The role of cationic functionality in dictating the binding to and disruption of lipid bilayers was investigated by a suite of biophysical techniques which indicated that primary ammonium groups more strongly complex to phospholipid headgroups as compared to QAS analogues.<sup>48</sup> A similar result was reported for poly(styrene) derivatives containing tertiary ammonium or QAS charges.<sup>49</sup> N.-L. Yang and co-workers explored antibacterial poly(acrylate) random copolymers with both primary and secondary ammonium groups in the side chains and found that the ratio of different cationic groups, as well as the charge density, plays a key role in dictating the antibacterial and hemolytic activities.<sup>50,51</sup>

Among the polynorbornene derivatives studied by the Tew group (Fig. 7), a variety of different cationic structures have been utilized. Primary ammonium cations were used first<sup>52</sup> and most frequently,<sup>53–55</sup> but pyridinium<sup>56</sup> and guanidinium groups<sup>54</sup> have also been utilized. In one study for example, Tew *et al.* showed that increasing the density of amine groups on each monomer unit enhances the efficacy of the polymers by decreasing hemolytic toxicity substantially.<sup>57</sup> Charge density was also examined by the Yan group in the context of geminized polyacrylates containing multiple same-centered QAS amphiphiles in each repeat unit. They found that the increased charge density per monomer unit enhanced antibacterial activity.<sup>58</sup>

Hedrick, Yang, and co-workers have extensively studied antibacterial polycarbonates (some example structures are

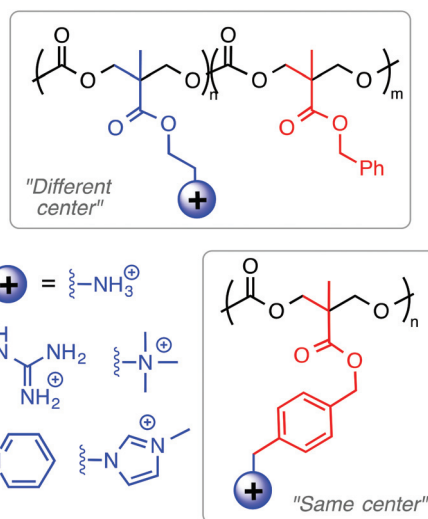


Fig. 8 Polycarbonates prepared by organo-catalyzed ROP have been structurally tuned to optimize the hydrophobic and cationic properties.

given in Fig. 8) with a variety of different QAS groups, including pyridinium and imidazolium.<sup>59,60</sup> These materials showed remarkable selectivity for *S. aureus* relative to human RBCs ( $HC_{50}/MIC > 250$ ) but were less effective against Gram-negatives and *C. albicans*. The J. Cai group studied amphiphilic polycarbonates (block and random) with primary ammonium functionality in the side chains, as compared to the QAS groups, and found potent selectivity against multidrug-resistant Gram-positives but not Gram-negatives.<sup>61</sup>

Besides high Lys content, AMPs are also enriched in Arg as a source of cationic charge. The guanidinium group in Arg is known to complex anionic phospholipids by a combination of Coulombic attraction and precisely orientated bidentate hydrogen-bonding interactions. Because anionic phospholipids are present on the outer leaflet of bacterial cell membranes, several groups have reported cationic amphiphilic polymers containing the guanidinium functional group as the source of cationic charge, instead of primary ammonium cations. Locock's group has shown that polymethacrylates with guanidinium cations were more active against *S. epidermidis* and *C. albicans* relative to primary ammonium-containing analogues.<sup>62</sup> Morgan and co-workers observed similar effects in guanidinium-functionalized poly(methacrylamide)s.<sup>63</sup> Recently, the Tew group has pioneered guanidinium-rich polymers as mimics of cell-penetrating peptides,<sup>64,65</sup> which is very interesting but outside the scope of this review.

Fernández-García and co-workers developed polymethacrylates bearing thiazolium and triazolium cations, which gave excellent antibacterial activity ( $4\text{--}8 \mu\text{g mL}^{-1}$ ) against *P. aeruginosa* and *S. aureus* with very high  $HC_{50}$  values ( $>5000 \mu\text{g mL}^{-1}$ ).<sup>66</sup> Tang and co-workers have pioneered the use of natural products in antimicrobial formulations. In one example,<sup>67</sup> they used tertiary amine groups quaternized with a derivative of the natural product rosin, which showed moderate antibacterial activity against *E. coli* and *S. aureus*. The same



Fig. 7 Examples from the polynorbornene design platform of Tew and co-workers. The identity and density of cationic charges, as well as the identity, relative amount, and spatial arrangement of the hydrophobic groups, have been extensively optimized.



group has shown that cationic polymethacrylates bearing cationic metallocene groups (cobaltocenium) kill multidrug-resistant bacteria and can also complex with small molecule antibiotic drugs in a synergistic formulation, paving the way for polymers to reinvigorate antibiotics that are presently considered obsolete.<sup>68</sup>

**“Same center” vs. “different center”.** HDPs contain cationic and hydrophobic groups that are present in different amino acid residues along the chain. In contrast, BACs contain a cationic group directly attached to a hydrophobic tail, as in surfactants or lipids. The former design strategy is sometimes called the “different center” amphiphilic approach, whereas the latter is called the “same center” approach. The first study to directly compare these two approaches was published in 2008 by the Sen group.<sup>69</sup> These authors creatively designed copolymers of alkyl quaternized vinyl pyridine and alkyl methacrylates. By independently varying the lengths of alkyl chains on the QAS and on the hydrophobic acrylic units, they disentangled the role of spatial arrangement of the amphiphilic groups. They concluded that a “different center” approach enhances the membrane disruption ability relative to the “same center” approach, but the latter is less prone to toxicity against human cells.<sup>69</sup>

Tew’s group has employed a strategy that is distinct from both the “same center” and “different center” motifs: in their polynorbornene derivatives, the cationic group and the hydrophobic group reside within the same repeat unit but are not directly attached to one another (Fig. 7). Rather, norbornenes with a hydrophobic group on one side and a cationic group on the other side are sometimes referred to as “facially amphiphilic” monomer units. The polymer chain composed of such units possesses conformational freedom, but is expected to adopt an overall facially amphiphilic conformation upon binding to biomembranes. It is reasonable to speculate that the monomer-level facial amphiphilicity may facilitate the formation of facially amphiphilic conformations at biointerfaces.

**“Facial” vs. “global” amphiphilicity.** Like any amphiphile, HDPs and their synthetic polymer mimics will tend to segregate into hydrophobic and hydrophilic domains in the presence of water. At the interface between an aqueous phase and the core of a phospholipid bilayer, these macromolecular amphiphiles have a propensity to “bury” their hydrophobic domains into the core of the membrane whereas the hydrophilic (and cationic) domains will prefer to interact with the

(anionic or zwitterionic) phosphate headgroups near the water interface, unsurprisingly. It has long been realized that membrane-active macromolecules ought to adopt “facially” amphiphilic conformations. That is, the hydrophobic segments should preferentially segregate to one side of the interface, whereas the cationic groups should be projected towards the other side (another way to express this idea is that the peptides are essentially rigid-rod Janus cylinders). The classical HDP Magainin-2 is a random coil in solution, but it adopts a facially amphiphilic helix upon binding to bio-membranes.<sup>70</sup> In efforts to mimic the pronounced facial segregation of cationic/hydrophobic residues, strategies have included designer peptide sequences,<sup>71–73</sup> rigid aryl-amide oligomers with stable facial conformations,<sup>74</sup> and homo-polymerization of monomers that are themselves facially amphiphilic.<sup>25</sup>

The Tew group presented a direct comparison of the “facially amphiphilic” monomer approach *versus* the “different center” approach, in polynorbornenes, and they showed a profound difference in activity and membrane-disruption ability in model membranes.<sup>75</sup> The Gellman group has argued that flexible random copolymers will naturally adopt “globally” amphiphilic conformations upon binding to biomembranes, without the need (or even the desire) for pre-programmed facial amphiphilicity at the monomer level or in terms of defined secondary structure.<sup>46</sup> Kuroda recently advocated a holistic view of antimicrobial polymer structures, emphasizing that these polymers should be designed to mimic the globally amphiphilic properties of the whole HDP molecule, instead of focusing on side chain identity.<sup>76</sup>

Considering the combined results from the “same center”, “different center” and “facially amphiphilic” approaches (Fig. 9), it is clear that each one requires independent optimization to yield antibacterial potency and low toxicity to RBCs. Overall, it is not clear whether any one strategy is fundamentally “better” than either of the other two: all three have yielded select examples with excellent activity profiles in terms of MIC (low  $\mu\text{M}$ ) and  $\text{HC}_{50}$  ( $>1000 \mu\text{M}$ ) *in vitro*. More studies that show a direct comparison between these approaches in terms of cytotoxicity and *in vivo* activity would perhaps clarify this issue further.

**Molecular weight.** HDPs typically contain  $\sim 10\text{--}50$  a.a., whereas biocidal polymeric disinfectants are often high molecular weight. Several groups have probed the role of chain length in dictating the antibacterial and hemolytic activities of



**Fig. 9** The relative spatial arrangements of cationic and hydrophobic groups in an antibacterial polymer generally fall into three categories: (A) “different-centered”, (B) “same-centered” and (C) “facially” amphiphilic structures. The approach in part (A) is evocative of the HDP primary structure, whereas (C) more closely mimics the typical secondary structure of a membrane-bound HDP. The approach in (B) appears similar to that of poly (BAC) disinfectants, but can exert activity similar to that of HDPs in some cases.



synthetic polymers. An exhaustive study by Kuroda *et al.* showed that random copolymers of methacrylate derivatives lost cell-type selectivity as the MW was increased.<sup>43</sup> Lienkamp *et al.* found that a thick peptidoglycan layer on model membranes (to mimic Gram-positives) can exert a molecular sieving effect: polynorbornene derivatives of MW  $\sim 3$  kDa diffuse through the barrier whereas higher MW analogues of  $\sim 50$  kDa were incapable of translocation across the peptidoglycan barrier.<sup>77</sup> Combined, these results reinforce the prudence of choosing shorter chain polymers as selective antibacterial compounds that mimic the short chain lengths that are typical in most AMPs.

**End groups.** For high molecular weight polymers, the influence of end groups on the amphiphilic balance is presumably negligible. HDP-mimetic antibacterial polymers are typically short oligomers, and thus the end group identity could play a pivotal role in dictating the properties. Controlled/“living” polymerization methods (RAFT and ATRP) were used to afford excellent control over end group fidelity in order to clarify the role of end groups in determining antibacterial and hemolytic activity. The earliest example from the Gellman group found that end groups on nylon-3 copolymers play a significant role, with more hydrophobic end groups causing greater hemolytic toxicity.<sup>45</sup> Locock studied polymethacrylates derived from a variety of RAFT agents with different R- and Z-groups (Fig. 10). The effect of R-group identity was rather pronounced: the neutral and hydrophobic isobutyronitrile group resulted in 20-fold greater hemolytic toxicity as compared to the cyanovaleic acid R-group. The antibacterial activity of a copolymer with a dodecylsulfanyl Z-group is substantially more potent

than one containing the less hydrophobic ethylsulfanyl Z-group.<sup>78</sup> Recently, the Whittaker group reported their studies on antibacterial poly(acrylate)s.<sup>79,80</sup> Compared to the poly(methacrylate) platform, these acrylate-based polymers are likely more flexible chains and thus are expected to display somewhat different activity profiles. Using ATRP, these authors varied the identity of cationic groups (primary, QAS, guanidine) as well as the end group derived from the initiator species. It was found that for very short oligomers of acrylate, the presence of a dodecyl end group dramatically enhanced antibacterial potency, in accord with Locock's results. These compounds could be thought of as very short diblock copolymers or as cationic surfactants that blur the distinction between small molecules and macromolecules. In a related study, Hedrick and Yang *et al.* showed that the incorporation of a cholesterol moiety to the end group of antibacterial polycarbonates enhanced the activity and also led to the formation of interesting disc-shaped nanoparticles in solution.<sup>81</sup> In summary, the role of end group identity should not be ignored in the synthesis of low MW oligomers as antibacterials.

## 2.2. Emerging SAR trends

**The missing ingredient: neutral, hydrophilic groups.** The first examples of HDP-mimetic antimicrobial polymers, and the vast majority of examples since, were optimized by adjusting the ratio, identity, and spatial arrangement of just two key components: a hydrophobic group and a cationic group. Indeed, these binary (co)polymer structures did capture some of the essential physicochemical features, as well as the concomitant antimicrobial activity, of HDPs. A very interesting study by G. C. L. Wong and co-workers revealed that the optimization of binary copolymers in fact gives structures with a larger number of cationic groups and greater overall hydrophobicity than the average HDP.<sup>82</sup> Thus, there seems to be a “missing ingredient” because it is impossible to further optimize a binary system if both of the two components are already too abundant. Of course, HDPs are *not* binary copolymers of hydrophobic and cationic residues; rather, they contain a diversity of amino acids that span a wide spectrum of physicochemical properties. In particular, the presence of a non-negligible fraction of *neutral, hydrophilic residues* in HDP sequences (Fig. 11) suggests that at least three components ought to be employed for molecular-level biomimicry. For this very reason, several groups have explored ternary copolymer variants of HDP-mimetic polymers containing neutral, hydrophilic groups such as hydroxyl, sugar, and grafted PEG chains. In certain cases, the third component indeed conferred a key advantage by lowering hemolytic toxicity while retaining (and in some cases even enhancing) antibacterial potency.

One of the earliest examples of a ternary antimicrobial polymer with neutral, hydrophilic groups, in addition to cationic amphiphilicity, was reported by Youngblood and co-workers in 2007 (Fig. 12).<sup>83</sup> They copolymerized 4-vinyl pyridine (4VP) with PEG methacrylate and then quaternized the pyridine groups with alkyl halides. It was previously known that quaternized poly(4VP) is inherently antibacterial but also



**Fig. 10** End group affects the antibacterial and hemolytic activity in polymethacrylates prepared by RAFT and polyacrylates prepared by ATRP.





**Fig. 11** Averaged composition of all 2903 AMPs in the database, percent incidence of each a.a. and their hydropathy indices. A substantial fraction (28%, or 40% including Gly) of the residues are neutral and hydrophilic. This portion of the AMP composition, previously neglected in binary systems, is mimicked effectively in ternary (co)polymer systems.

highly hemolytic. Incorporating the PEGylated comonomer reduced the hemolytic toxicity and did not abrogate the antibacterial properties. Thus, by including a third component in the design, these authors effectively demonstrated that a nominally biocidal polymer can be modified to confer cell-type selectivity.

Numerous other examples (Fig. 12) have since shown that the same design principle can be applied to polymers of

norbornenes,<sup>84</sup> methacrylates,<sup>85</sup> and nylon-3 type materials.<sup>86</sup> PEG has frequently been employed for such purposes, but hydroxyl substituents (which mimic Ser residues),<sup>87,88</sup> sugars,<sup>89,90</sup> and zwitterionic moieties<sup>84</sup> also give comparable results. In our opinion, these ternary systems are likely promising candidates for continued study.

**Chain architecture.** The Boyer and Wong groups have elegantly studied the role of polymer chain architecture, sequence, and chain length on the antibacterial and hemolytic activities of functional poly(acrylate)s, poly(acrylamide)s, and their copolymers, prepared by RAFT methods.<sup>91</sup> They synthesized polymer single chain nanoparticles (SCNPs) that displayed excellent activity. Random coil conformations in polymer SCNPs clearly lack the structural precision of a folded protein or peptide, but they nevertheless represent a crude version of polymer folding that clearly has a large impact on biological activity. Very recently, the same groups utilized ternary copolymer formulations (cationic, hydrophobic, and neutral/hydrophilic) to directly compare linear random *versus* block copolymers as well as hyper-branched analogues.<sup>92</sup> It was found that the hyper-branched polymers outperformed the linear copolymers in terms of cell-type selectivity.

G. G. Qiao and co-workers used PAMAM dendrimers to initiate the ring-opening polymerization of *N*-carboxyanhydrides to yield particles decorated with random copolymers of Lys and Val (Fig. 13).<sup>93</sup> These nanoparticles exhibited potent activity against Gram-negative bacteria, were non-toxic to human cells, and did not induce drug resistance. It is critically important for *in vivo* applications that serum proteins do not abrogate the activity of the polymers, although some interaction between a polycation and a protein in the media is likely unavoidable. The dendrimer-based nanoparticles were remarkably active even in the presence of simulated body fluid and blood serum, which mimic the *in vivo* environment. Interestingly, the presence of added divalent salt slowed the kinetics of *E. coli* outer membrane permeabilization, which



**Fig. 12** Ternary amphiphilic polymers containing hydrophobic, cationic, and neutral/hydrophilic groups (hydroxyl, sugar, PEG, and zwitterionic). The inclusion of the third component reduces the hemolytic toxicity without sacrificing antibacterial potency.





**Fig. 13** Antibacterial nanoparticles by ring-opening polymerization initiated by a PAMAM dendrimer core. This unique chain architecture influences the mechanism of activity relative to linear chains. Reproduced with permission from ref. 93, copyright 2016, Springer Nature.

suggests that ion exchange may play an important role in their mechanism of action.

Antibacterial nanostructures composed of spherical and rod-shaped graft copolymers that display quaternized poly(vinyl pyridine)s showed a strong dependence of activity on chain architecture.<sup>94</sup> Interestingly, the smaller spherical-shaped nanostructures exhibited stronger antibacterial activity as compared to rod-like graft copolymers and linear polymer chains.

Combined, the results from these studies strongly support the notion that polymer chain microstructure, chain architecture, and nanostructures are important determinants of biological activity. Efforts to examine the role of sequence, tacticity, regioregularity, and chain architecture appear worthwhile and ought to be the focus of more attention in the future.

**Cationic spacer length.** Inspired by the so-called “snorkel” effect in trans-membrane peptide helices,<sup>95,96</sup> Palermo and Kuroda<sup>97</sup> hypothesized that tuning the length of the side chain spacer group that connects the side chain primary ammonium cation to the polymer backbone would enable the control of the depth of penetration of the non-polar backbone into the hydrophobic membrane core, thus tuning the antibacterial activity (Fig. 14).<sup>97</sup> Indeed, the MIC and HC<sub>50</sub> values were sensitively dependent on spacer length. The elongation of the C<sub>2</sub> spacer (amino-ethylmethacrylate) to a C<sub>4</sub> spacer (amino-butylmethacrylate) enhances the antibacterial activity without substantially increasing the hemolytic toxicity. Further elongation to a C<sub>6</sub> spacer dramatically aggravated the hemolysis. MD simulations shed light on the conformations of these polymers in the membrane-bound state as a function of spacer arm length. Indeed, the polymer with a C<sub>4</sub> spacer group adopted a pronounced facially amphiphilic conformation upon membrane binding. N.-L. Yang and co-workers have employed a related “spacer arm” strategy to optimize the antibacterial activity of closely related poly(acrylate)s. In one particularly illustrative comparison, they showed that a polymer with C<sub>6</sub> spacers exerts dramatically different activity when directly compared to the isomeric polymer containing a C<sub>2</sub> spacer and



**Fig. 14** The length of the spacer arm that connects a cation to the polymer backbone is a key design parameter.

a C<sub>4</sub> same-centered tether.<sup>51</sup> These studies strongly support the notion that the spacer arm plays a pivotal role in modulating the polymer–membrane interactions and concomitant activity.

**Self-immolative antimicrobial polymers.** The majority of the synthetic antimicrobial polymers are non-biodegradable, which could limit their application due to possible long-term toxicity *in vivo*, even if they appear to be non-toxic by short-term *in vitro* assays.<sup>98,99</sup> Biodegradable antimicrobial polymers based on polyesters,<sup>99,100</sup> polyurethanes,<sup>101,102</sup> polycarbonates<sup>59,60</sup> and networks cross-linked with acetal functionality<sup>103</sup> have been reported to fill this gap. These polymers are hydrolytically or enzymatically cleaved at random sites along the backbone, thus giving a degradation profile that is controlled by the chemical structure of the linkages and the solvent accessibility of the microenvironment.<sup>104</sup> Polymers that can degrade in a specifically *triggered* manner may confer some advantages relative to passive degradation.<sup>105</sup> A novel and interesting class of materials that undergo triggered depolymerization are the so-called “self-immolative” polymers (SIMPs), designed to unzip into a monomer upon the cleavage of their  $\omega$ -end-groups.<sup>106–115</sup> SIMPs have been utilized in drug delivery,<sup>116–118</sup> biosensors,<sup>119</sup> microfluidics<sup>120</sup> and dynamic plastics<sup>121</sup> but did not find use in the context of antimicrobials until very recently.

Ergene and Palermo developed the first generation of self-immolative antibacterial polymers<sup>122</sup> (Fig. 15). They synthesized poly(benzyl ether)s with pendant allyl side chains and silyl ether end-caps (responsive to fluoride ions),<sup>123</sup> following the route reported by Phillips and co-workers. They further modified side chains with cysteamine HCl *via* photo-initiated thiol–ene chemistry. Cationic poly(benzyl ether)s bearing primary ammonium groups exerted potent, rapid and broad-spectrum antibacterial activity, but were also highly hemolytic.





ation, only a few reports on alternating antibacterial copolymers exist in the literature. It is straightforward to prepare alternating copolymers by step-growth, but these materials will necessarily have quite broad chain length dispersity. An outstanding example by the Sampson group<sup>127</sup> showed that the ROMP of 1-substituted cyclobutenes and cyclohexenes affords alternating copolymers with precise spacing between functional groups (Fig. 17). They found that alternating sequences of cationic and hydrophobic groups outperformed random copolymer analogues in terms of potent antibacterial activity and minimal hemolytic toxicity, with better than 100-fold selectivity for the best examples. Interestingly, increasing the hydrophobicity of the groups between the cations did not dramatically aggravate the hemolytic toxicity, as typically seen with other copolymer mimics of HDPs. Related advances have been made by the Haldar group using alternating copolymers of maleimide and isobutylene.<sup>128</sup> These findings suggest that alternating sequence is a prudent and perhaps under-appreciated design strategy.

Very recently, Perrier and co-workers undertook a detailed study of sequence control in antimicrobial poly(acrylamide)s bearing hydrophobic isopropyl and cationic aminoethyl side chains.<sup>129</sup> They prepared statistical, diblock, and multiblock linear copolymers. The antibacterial and hemolytic activities are significantly impacted by the sequence distribution; the MIC against *P. aeruginosa* decreased from 1000  $\mu\text{g mL}^{-1}$  for the statistical copolymer to 32  $\mu\text{g mL}^{-1}$  for the diblock and 8  $\mu\text{g mL}^{-1}$  for the heptablock whereas all three samples were

non-hemolytic. Thus, one may conclude that the clustering of hydrophobic side chains into small domains along the chain, as in a multiblock architecture, is a prudent strategy to enhance the antibacterial activity without affecting the toxicity to human cells.

**Tacticity.** The classical example used to illustrate the importance of stereoregularity is the glass transition temperature  $T_g$  of PMMA, which depends sensitively on tacticity ( $T_g = 124\text{ }^\circ\text{C}$  for syndiotactic *versus*  $42\text{ }^\circ\text{C}$  for isotactic). The extent to which tacticity matters in the context of antibacterial polymers remains unexplored, but one might reasonably speculate that the chain conformational flexibility/rigidity in aqueous solution would play a role in modulating the interaction of polymer chains with biointerfaces. Even though precise control of stereochemistry is not required for activity, the physicochemical properties of the polymer chains likely do depend on tacticity. To our knowledge, no study has ever quantified the role of tacticity in determining the antibacterial and hemolytic activities of a polymer.

#### 2.4. The real deal: activity *in vivo*

The field of antibacterial polymers initially focused on the optimization of structures to minimize MIC and maximize  $\text{HC}_{50}$  values *in vitro*. Having identified many promising examples from a large pool of candidates, the next logical step is to quantify the activity *in vivo* using various animal models. This is a non-trivial advance because polymers that perform well *in vitro* may be inactivated *in vivo* by serum proteins, platelet adhesion, or other non-specific interactions. Nevertheless, a growing number of recent studies have shown encouraging results, which collectively represent a tremendous vertical advance of the field.

Thoma *et al.* showed that a topical formulation based on cationic homopolymers of aminoethyl methacrylate eradicated *S. aureus* infection in a mouse nasal infection model (Fig. 18).<sup>130</sup> These low MW polymers (3–10 kDa) are bactericidal to *S. aureus* at low micromolar concentrations even in the presence of serum *in vitro* and are non-toxic to mammalian cells (hRBCs, HEP-2 and COS-7 cell lines). They also demonstrated eradication efficacy *in vivo* that was significantly better than the small molecule antibiotic drug mupirocin. Further, it



Fig. 17 Alternating copolymers as platforms for antibacterial polymers.



Fig. 18 Cationic polymethacrylates eradicated *S. aureus* in a mouse nasal infection model. Reproduced from ref. 130, ACS AuthorChoice (Open Access).



was confirmed that the polymers do not induce resistance after 15 sub-inhibitory exposures. These results all strongly suggest that such polymers are good candidates for topical applications to cure *S. aureus* infections.

Haldar and co-workers have reported the *in vivo* efficacy of various HDP-mimetic copolymers<sup>131</sup> and small molecules.<sup>132</sup> Alternating copolymers of maleimide and isobutylene bearing a same-centered QAS amphiphile in the side chains were effective against *A. baumannii* in a burn infection mouse model *via* topical application.<sup>131</sup> The concentration of viable bacterial cells in the burn was reduced from  $\sim 10^{10}$  to  $10^7$  CFU mL<sup>-1</sup> upon treatment with 50 mg kg<sup>-1</sup> of polymer, which is comparable to the efficacy of the antibiotic minocycline at the same dosage. When both the polymer and minocycline were used in combination, the *A. baumannii* concentration was further reduced to  $\sim 10^4$  CFU mL<sup>-1</sup>. The same group showed that cationic and amphiphilic small molecules composed of a single fatty acid chain conjugated to two equivalents of lysine (+4 charge at pH 7.4), which could be viewed as a minimalist design to mimic the HDP structure, successfully eradicated MRSA biofilms from infected mouse skin.<sup>132</sup>

P. X. Ma and co-workers reported the *in vivo* activity of antimicrobial bottle-brush polymethacrylates grafted to PDMS in a subcutaneous rat infection model.<sup>133</sup> Random copolymers of lysine and phenylalanine prepared by ROP were conjugated to heterobifunctional PEG methacrylate and then subjected to surface-initiated/UV-triggered polymerization to yield the bottle-brush grafted surface. These coated PDMS samples, alongside bare PDMS, were inoculated with  $10^7$  CFU of *E. coli* and then implanted into rats subcutaneously. After 5 days, the excised implants coated with the polymer contained only 0–10 CFU (a 6- or 7-log reduction) whereas the bare PDMS surface was still host to  $\sim 10^5$  CFU. Visually, the bare PDMS surface side of the wound showed extensive bacterial infection whereas the coated side appeared to be normal tissue with no sign of infection. Thus, these polymethacrylate-*graft*-(PEG-poly(Lys-*block*-Phe)) materials substantially reduced the implant-associated infection under rat skin, representing a major advance for implant materials.

The G. G. Qiao group reported *in vivo* efficacy of their antimicrobial polymer nanoparticles in a mouse peritonitis model. They found that the concentration of *A. baumannii* in the peritoneal wash was dramatically reduced (a 5-log reduction) upon dosing with antibacterial dendrimers, which was similar to the efficacy of the antibiotic drug imipenem.<sup>93</sup>

The cationic and amphiphilic polycarbonates of Hedrick and Yang showed high *in vivo* efficacy against MRSA by systemic application in immunosuppressed mice (Fig. 19).<sup>134</sup> The mice were injected with  $4 \times 10^8$  CFU mL<sup>-1</sup> of MRSA, which was determined to be a lethal dose. Amazingly, the polymer outperformed the antibiotic vancomycin in terms of reduction in MRSA concentration (to  $10^3$  CFU mL<sup>-1</sup>) over a time period of 4 and 8 hours post-injection. The effective dose of the polymer was  $\sim 2$  mg L<sup>-1</sup> against MRSA whereas the lethal dose against the mice was  $\sim 20$  mg L<sup>-1</sup>, suggesting a 10-fold selectivity. After 7 days post-infection, 14 out of 15 mice that received the



Fig. 19 Rapid reduction in the concentration of MRSA cells in the bloodstream of a mouse was achieved with antibacterial poly(carbonate)s compared to the antibiotic drug vancomycin. Reproduced with permission from ref. 134, copyright 2015, Wiley-VCH.

polymer treatment survived whereas none of the control mice (injected with MRSA in PBS buffer) survived. Hence, this breakthrough study provides very strong evidence that cationic, amphiphilic polymers can cure infection in mice when administered systemically.

Overall, these early-stage *in vivo* studies collectively demonstrate that HDP-mimetic polymers possess great promise as therapeutic agents to combat drug-resistant bacterial infections. Still, the relative paucity of survival studies at this early stage indicates a continued need for further progress.

### 3. Mechanism of action

#### 3.1. HDP mechanism

Since the 1980s, strongly cationic peptides have been known to disrupt the integrity of bacterial membranes.<sup>12,135</sup> At the simplest level, the peptide-membrane interaction is understood in terms of electrostatic attraction between the anionic bacterial cell envelope and cationic residues in the peptide, as well as partitioning of hydrophobic residues into the nonpolar membrane core. The putative mechanism of HDP action is often described in molecular detail by the Shai-Matsuzaki-Huang (SMH) model.<sup>7,136,137</sup> As a classical example, Magainin-2 adopts a facially amphiphilic  $\alpha$ -helix at the membrane interface upon binding, and then cooperatively forms transient pentameric pores across the bilayer.<sup>138</sup> An abundance of evidence supports some aspects of the SMH model for a wide variety of HDPs.<sup>4,139–142</sup> Nonetheless, HDPs may exert mechanisms other than (or in addition to) membrane disruption, including immunomodulatory effects, metabolic inhibition, and programmed cell death.<sup>5,143,144</sup> Here, we review mechanistic studies that probe HDP-mimetic polymers, with a sole focus on the membrane disruption aspects. Because the polymers possess the critical features of HDPs, and show similar antibacterial activity, it is widely hypothesized that they act by related mechanisms of action, despite their molecular heterogeneity and lack of defined secondary structure.



### 3.2. Liposomes as model membranes

Liposomes are frequently employed as simplified model membrane systems to probe the interaction of polymers with phospholipid bilayers using a wide range of physicochemical methods such as spectroscopy, microscopy and thermal analysis.

A classical experiment to evaluate membrane permeabilization induced by polymers is to monitor the release of a fluorescent dye entrapped within a liposome upon mixing with a dilute polymer solution (Fig. 20). A water-soluble fluorescent dye, such as carboxyfluorescein, calcein or sulforhodamine-B, is entrapped in large unilamellar vesicles (LUVs) above the self-quenching concentration (typically 50 mM). Upon the disruption of the bilayer barrier function, the dye is released into the dilute solution and fluorescence intensity is markedly recovered.<sup>43,56,145</sup> Phospholipid composition is chosen to mimic various cell types. For instance, the mixture of anionic phosphatidylglycerol (PG) and zwitterionic phosphatidylethanolamine (PE) is evocative of *E. coli* membranes whereas phosphatidylcholine (PC) and cholesterol are used as a proxy for mammalian cell membranes.<sup>146</sup> Despite its simplicity, dye leakage provides a useful demonstration of the inherent membrane-disrupting ability of a polymer, which may contribute to the mechanism of antibacterial and hemolytic action (at least in part).<sup>147,148</sup> This technique has been employed for polynorbornenes,<sup>56</sup> polymethacrylates,<sup>43</sup> and nylon-3 copolymers.<sup>147</sup> The fundamental basis for cell-type selectivity can be observed by comparing dye leakage from liposomes that mimic anionic bacterial membranes (e.g. POPE/POPG) and those that mimic zwitterionic mammalian cell membranes (POPC/chol). Membrane permeabilization by HDPs depends on the lipid

charge as well as intrinsic curvature.<sup>149</sup> Gellman *et al.* showed that their HDP-mimetic nylon-3 polymer selectively disrupts lipid membranes depending on the lipid composition; the polymer efficiently disrupts liposomes of DOPG/DOPG (mimicking *E. coli* membranes), but not those of POPC (mimicking mammalian cell membranes).<sup>147</sup> These data suggest that optimized formulations of HDP-mimetic polymers are inherently more active against membranes that possess anionic charge and negative curvature. Fluorescence lifetime assays can distinguish between *all-or-none* leakage (a fraction of the vesicles released all of their contents) *versus graded* leakage (all vesicles released a portion of their contents). The type of mechanism depends on polymer hydrophobicity: a nylon-3 derivative consisting solely of cationic units induces *all-or-none* leakage, whereas a cationic/hydrophobic copolymer exhibits graded leakage.<sup>147</sup>

**Dye-labeled polymer.** The binding affinity of the antimicrobial polymers to lipid bilayers can be estimated based on dye labeling with dansyl, which is sensitive to the polarity of its microenvironment.<sup>150</sup> Dansyl-labeled antimicrobial polymethacrylates exhibit a fluorescence intensity increase, and a blue shift, upon the transfer from aqueous solution to a non-polar membrane environment.<sup>43</sup> This characteristic enables fluorescence titration to obtain a binding isotherm for the quantification of the polymer affinity to membranes. Kuroda *et al.* have used such polymers to study the role of hydrophobicity and lipid composition in the polymer–membrane binding event. The dansyl-modified polymer also clarified the effect of the cationic group structure on the membrane binding; the polymer with primary ammonium side chains displayed higher binding affinity to POPC liposomes relative to tertiary and quaternary analogues, which correlated with the differences in their hemolytic activities.<sup>151</sup>

**Giant vesicles.** Micron scale “giant” unilamellar vesicles (GUVs) are approximately cell-sized and thus enable the optical visualization of membrane dynamics. Microscopic observation of calcein-entrapped GUVs upon the addition of Magainin-2 showed an induction time, followed by quantitative leakage of the entrapped dye without any morphological changes in the membrane, suggesting stochastic pore formation.<sup>152</sup> GUV studies have been carried out for a wide range of other HDPs<sup>153–155</sup> as well as synthetic polymers.<sup>156</sup> The addition of phenylene ethynylene antimicrobial oligomers to a giant vesicle composed of *E. coli* extract lipid induced the leakage of a small (1 kDa) marker molecule whereas a large (19 kDa) marker molecule was retained in the vesicle, suggesting a size exclusion effect of defined pore size.<sup>156</sup>

**SAXS.** G. C. L. Wong and coworkers have extensively studied the impact of HDPs and their synthetic mimics against lipid membranes using small-angle X-ray scattering (SAXS), a powerful method to probe topological changes in lipid membranes induced by antimicrobials (Fig. 21). They demonstrated that the cationic phenylene ethynylenes induced the transition of small unilamellar vesicles into a non-lamellar inverted hexagonal phase with water channels ~3.4 nm in diameter.<sup>156</sup> Cationic polymethacrylate derivatives were found to induce the



Fig. 20 Schematic of the liposome dye leakage induced by a cationic amphiphilic polymer.





Fig. 21 SAXS data (a,b) for a lipid bilayer containing antimicrobial polymethacrylates and corresponding real space model (c). Reproduced with permission from ref. 157, copyright 2013, American Chemical Society.

transformation of the membrane to a bicontinuous cubic phase with a negative Gaussian curvature.<sup>157</sup> Additionally, the presence of phosphatidylethanolamine, which has an intrinsically negative curvature, enhances the formation of the non-lamellar structure with a negative Gaussian curvature upon the addition of antimicrobials.

**DSC.** Epand *et al.* have utilized differential scanning calorimetry (DSC) to show that the phase transition behavior of lipid membranes is sensitively influenced by the incorporation of antimicrobial peptides,<sup>158,159</sup> synthetic oligo-acyl-lysine<sup>160</sup> and polymers.<sup>161</sup> These compounds induce lateral phase separation in the membranes by the clustering of the anionic lipids, which results in phase-boundary defects wherein the barrier function is severely curtailed. By this mechanism, antimicrobials may permeabilize the membrane without the formation of well-defined pores. Further support for this unique mechanism comes from MD simulations (*vide infra*).

**SFG.** The Chen group has employed sum frequency generation (SFG) in combination with a supported lipid bilayer to examine membrane-bound antimicrobial molecules.<sup>162–164</sup> For example, they examined the interaction of antimicrobial polymethacrylates with a deuterated lipid bilayer. Strongly hydrophobic copolymers markedly attenuate the C–D stretching SFG

signal from the outer and inner leaflets of the bilayer, which implies disruption of bilayer ordering. Additionally, the C–H stretching signal that originates from the polymer revealed the orientation of hydrophobic sidechains aligned with the lipid bilayer.

**NMR.** The Ramamoorthy group has used solid-state <sup>31</sup>P NMR to study membrane deformation induced by HDPs and their synthetic mimics. The <sup>31</sup>P signal of the phospholipid head group is highly sensitive to orientation, phase, and complexation with cations.<sup>165,166</sup> Thus, the spectrum of a POPC bilayer is markedly altered by the incorporation of antimicrobial polymers.<sup>151</sup> While the majority of lipids maintain their lamellar order with a characteristic peak, the appearance of a new peak corresponding to the lipid head group oriented perpendicular to the membrane suggests that pore formation is indeed induced by the polymer.

### 3.3. Molecular dynamics simulation

All-atom molecular dynamics simulations are rapidly emerging as a very powerful tool to probe polymer–lipid interactions in precise detail at the molecular level (Fig. 22). The MD simulation of an antimicrobial polymethacrylate binding to a phospholipid bilayer revealed that primary amines in the side chains associate with phosphate head groups in the lipids, whereas the insertion of the polymer into the membrane is mediated predominately by the hydrophobic effect, as expected.<sup>167</sup> Most interestingly, it was found that the comonomer sequence plays a pivotal role in dictating the overall shape of the chain molecule, which implies that sequence control may indeed be an important molecular design consideration that remains underexplored to date.

Also in the context of antimicrobial polymethacrylates, MD was employed to clarify the role of the cationic spacer group (*i.e.*, the length of the carbon chain connecting the cationic ammonium group to the backbone). Interestingly, the elongation of the spacer group promotes the deeper insertion of the



Fig. 22 Snapshots from an MD simulation of a polymethacrylate on a lipid bilayer. Reproduced with permission from ref. 170, copyright 2017, RSC Publishing.



polymer backbone into the membrane core, in analogy to the “snorkel effect” in transmembrane peptides.<sup>95,168,169</sup> The polymers with intermediate spacer length ( $C_4$ ) adopt a facially-amphiphilic structure upon binding to the membrane, in which cationic and hydrophobic sidechains are segregated to opposite faces.<sup>97,170</sup> This is a remarkable observation reminiscent of the secondary structure in membrane-bound Magainin-2, despite the complete lack of structural perfection in random copolymethacrylates. Related MD simulations also showed the clustering of negatively-charged lipids into distinct domains upon polymer binding, which resulted in lipid coarsening and bimodal membrane thickness.<sup>171</sup> Such “charge clustering” in the membrane is known to produce a packing defect, wherein the permeability barrier is compromised, in corroboration with the experimental DSC data from Epan.<sup>159</sup>

### 3.4. Seeing is believing

Direct observation of bacterial cell membranes in the presence of an antimicrobial polymer provides the most convincing evidence that their mechanism of action involves the membrane as a target, albeit with less quantitative molecular-level understanding. Such visualization has been done extensively by TEM, which can reveal detailed morphological changes in the membrane structure upon exposure to antimicrobial polymers (Fig. 23). For example, Yang and Hedrick reported biodegradable nanostructures by the self-assembly of cationic and hydrophobic triblock copolycarbonates.<sup>134</sup> Comparative TEM images of methicillin-resistant *S. aureus* (MRSA) in the presence of a polymer (slightly above the MIC) showed significant damage to the cell membrane accompanied by the leakage of cell contents, whereas the control images show smooth, healthy cell surfaces.<sup>98</sup> It is apparent from the images that cell lysis occurs by an all-or-none mechanism. The authors hypothesize that micelles of these triblock copolymers, which are attracted to the anionic cell surface by electrostatic inter-

actions, may inhibit cell wall synthesis due to their steric bulk, leading to membrane disruption. Alternatively, the micelles might permeabilize the cytoplasmic membrane as a result of electroporation or the sinking raft model, which could also lead to cell death.

In TEM images, the shape of the bacterial cells can be visualized, but the polymers themselves are essentially invisible. In order to directly track the fate of the polymers *in vitro*, Reynolds and Qiao prepared dendrimers of an antimicrobial polymer tagged with AlexaFluor-488 (green) and employed super-resolution fluorescence 3D structured illumination microscopy (3D-SIM) for visualization (Fig. 24).<sup>93</sup> *E. coli* cells labeled with an optically orthogonal membrane dye (red) were mixed with an AF488-tagged polymer, above and below the lethal dose, to investigate the polymer particle localization in relation to the bacterial membrane.

Whereas various microscopy techniques have been employed to study polymer–membrane interactions previously,<sup>172,173</sup> this elegant study was the one of the first that used super-resolution optics to unambiguously confirm the co-localization of an antimicrobial particle on a bacterial cell membrane at high resolution.<sup>93</sup> Below the lethal dose (1/2 MBC), these particles sparsely populate the outer surface of the *E. coli* cells, whereas the surface is nearly saturated with particles at the MBC. At double the MBC, particles aggregate on and penetrate into the cells, leading to a clustering of lipids in the membrane and clear loss of membrane integrity. It is clearly seen in the red channel that the membrane lipids are clustered into dense domains, leaving behind patchy dark regions (Fig. 24, panels e–h). Also, aggregates of red lipid can be seen in the surrounding media, suggesting perhaps that micelles or vesicles of lipid have been shed from the cell surface. Based on these observations, the authors speculated that the polymers initially adhere to the outer membrane (OM) due to electrostatic attraction, which might destabilize the OM and permit translocation to the cytoplasmic membrane (CM). In support of this view, they demonstrated that the polymers avidly bind LPS, a major component of the OM. The polymer particles did not induce pore formation in model membranes, but did cause unregulated ion movement and dissipation of membrane potential. Finally, the polymers induced a programmed cell death (PCD)<sup>144</sup> pathway that is triggered under stressful conditions such as membrane perturbation, although the polymer retains the antibacterial activity in the presence of an inhibitor that blocks this PCD pathway. Hence, the authors conclude that PCD is not a requisite mechanism for antibacterial activity but that it may occur as a supplemental, though not required, alternative mechanism of bacterial lethality.<sup>93</sup>

It is important to note that nanoparticles composed of antibacterial polymers may exert subtly different mechanism(s) of action as compared to the corresponding individually solvated polymer chains. These nanoparticles contain multiple chains that are co-localized in space and are thus inherently more locally concentrated upon membrane binding. By comparison, individual chains must cooperatively bind membranes in order to achieve local concentrations comparable to that of the



Fig. 23 TEM images of intact MRSA cells (left) and damaged cells upon exposure to an inhibitory dose of antibacterial polycarbonate (right), at two magnifications. Reproduced with permission from ref. 98, copyright 2011, Springer Nature.





**Fig. 24** 3D-SIM superresolution fluorescence images of an antibacterial polymer particle (PAMAM dendrimer with surface-initiated random copolymers of Lys and Val) with *E. coli* cells. The particles are tagged with AF488 (green) and the *E. coli* cell membrane is dyed with FM4-64FX (red). Control *E. coli* (a), polymer at half the MBC (b), at the MBC (c–e), and at twice the MBC (f–h). Reproduced with permission from ref. 93, copyright 2016, Springer Nature.

nanoparticles. In addition, non-specific binding to serum proteins, kinetics of degradation, details of membrane permeabilization and/or translocation and the intracellular fate of the nanoparticles may all differ from those of their constituent polymeric chains. While the bulk of prior mechanistic work has been done on individual polymer chains, many of the most cutting-edge materials currently being explored for *in vivo* application are increasing focused on nanoparticles and nano-assemblies of antibacterial polymers. Thus, mechanistic studies on polymer nanoparticles, both *in vitro* and *in vivo*, are still urgently needed at the forefront of this field. In summary, the literature is replete with evidence, ranging from the molecular level to the nano- and micro-scales, that HDP-mimetic antibacterial polymers indeed act, at least in part, by a mechanism involving the disruption of the bacterial cell membrane barrier function.

## 4. Conclusions

The fascinating discovery of HDPs and the development of polymeric disinfectants in the 1980s set the stage for molecular level biomimicry using synthetic polymers. Since the ground-breaking studies of DeGrado and others in the early 2000s, the field of peptide-mimetic antimicrobial polymers has continuously expanded. By focusing on the key physicochemical structural determinants of activity in host defense peptides, some well-established design rules are now in place: (1) a finely tuned balance of hydrophobic and cationic moieties, (2) protonated primary amine (or guanidine) groups as the source of cationic charge, and (3) relatively low MW, which tend to confer the desired bioactivity profiles. Further fine-tuning of the activity profiles has been achieved by incorporat-

ing neutral/hydrophilic components, and by exploring the roles of chain sequence and architecture. PEGylation in particular is a proven method to alleviate hemolytic toxicity without sacrificing antibacterial potency. Molecular platforms for optimization over the past ten years have included the polynorborene class of Tew and co-workers, the polymethacrylates of Kuroda's and several other groups, and the nylon-3 class spearheaded by Gellman's group.

Ever more diverse and creative variants continue to appear in the literature. Hyperbranched polymers, dendrimers, and single chain nanoparticles have shown excellent promise as antibacterials with low toxicity. Other contributions have included the demonstration of *in vivo* efficacy, optimization of biodegradable platforms, the self-immolative examples, and the use of renewable feedstocks, as just a few examples. These recent efforts, and many others, continue to invigorate the rapidly expanding multidisciplinary subject. Still, much work remains to be done. Precise control structural features including comonomer sequence, tacticity, chain length, and compositional dispersity have not been exhaustively correlated with activity. While many excellent studies on the interaction between polymers and model lipid bilayers provided a valuable foundation of knowledge, a complete understanding of the mechanism of action exerted *in vivo* by this increasingly diverse class of biomacromolecules is still urgently needed.

In terms of practical applications, antimicrobial polymers have found use in the solid state as coatings/surfaces on consumer products and medical devices, food-packaging materials, and textiles.<sup>174</sup> Numerous HDP-inspired antimicrobial *peptides* have completed or are currently in clinical trials,<sup>175</sup> but thus far no synthetic *polymer* HDP-mimic has reached such a benchmark, to our knowledge. Potential pitfalls abound in the development of any new drug and anti-



microbial polymers are certainly no exception: pharmacokinetics, long-term toxicity, biodistribution, possible septicemia caused by bacterial cell lysis (if that is the mechanism of action), and potential immunogenicity arising from a polymer's degradation by-products (if it is degradable) are all challenges that will likely require sustained efforts to surmount. Although synthetic polymers possess heterogeneity in chain length, composition, and stereochemistry, we can see no fundamental reason why these features inherently prohibit synthetic polymers from exploration in the clinical setting. Moreover, continued advances in the precision control of polymer synthesis may soon circumvent such concerns altogether. In closing, we look forward to continued growth and future developments in this exciting interdisciplinary field of polymer science.

## Conflicts of interest

There are no conflicts to declare.

## References

- H. W. Boucher, G. H. Talbot, J. S. Bradley, J. E. Edwards, D. Gilbert, L. B. Rice, M. Scheld, B. Spellberg and J. Bartlett, *Clin. Infect. Dis.*, 2009, **48**, 1–12.
- R. E. W. Hancock and H. G. Sahl, *Nat. Biotechnol.*, 2006, **24**, 1551–1557.
- R. I. Lehrer and T. Ganz, *Curr. Opin. Immunol.*, 1999, **11**, 23–27.
- M. Zasloff, *Nature*, 2002, **415**, 389–395.
- K. A. Brogden, *Nat. Rev. Microbiol.*, 2005, **3**, 238–250.
- K. A. Brogden, M. Ackermann, P. B. McCray and B. F. Tack, *Int. J. Antimicrob. Agents*, 2003, **22**, 465–478.
- Y. Shai, *Biochim. Biophys. Acta, Biomembr.*, 1999, **1462**, 55–70.
- K. Matsuzaki, *Biochim. Biophys. Acta, Rev. Biomembr.*, 1998, **1376**, 391–400.
- M. L. Juba, D. K. Porter, E. H. Williams, C. A. Rodriguez, S. M. Barksdale and B. M. Bishop, *Biochim. Biophys. Acta, Biomembr.*, 2015, **1848**, 1081–1091.
- G. E. Fantner, R. J. Barbero, D. S. Gray and A. M. Belcher, *Nat. Nanotechnol.*, 2010, **5**, 280–285.
- M. Wilmes, B. P. A. Cammue, H. G. Sahl and K. Thevissen, *Nat. Prod. Rep.*, 2011, **28**, 1350–1358.
- M. Zasloff, *Proc. Natl. Acad. Sci. U. S. A.*, 1987, **84**, 5449–5453.
- G. S. Wang, X. Li and Z. Wang, *Nucleic Acids Res.*, 2016, **44**, D1087–D1093.
- G. S. Wang, X. Li and Z. Wang, *Nucleic Acids Res.*, 2009, **37**, D933–D937.
- Z. Wang and G. S. Wang, *Nucleic Acids Res.*, 2004, **32**, D590–D592.
- Y. Lan, J. T. Lam, G. K. H. Siu, W. C. Yam, A. J. Mason and J. K. W. Lam, *Tuberculosis*, 2014, **94**, 678–689.
- A. Won, M. Khan, S. Gustin, A. Akpawu, D. Seebun, T. J. Avis, B. O. Leung, A. P. Hitchcock and A. Ianoul, *Biochim. Biophys. Acta, Biomembr.*, 2011, **1808**, 1592–1600.
- K. Hall, E. Porter, N. Umezawa, S. H. Gellman and M. Aguilar, *J. Pept. Sci.*, 2004, **10**, 175–175.
- T. L. Raguse, E. A. Porter, B. Weisblum and S. H. Gellman, *J. Am. Chem. Soc.*, 2002, **124**, 12774–12785.
- E. A. Porter, B. Weisblum and S. H. Gellman, *J. Am. Chem. Soc.*, 2002, **124**, 7324–7330.
- W. Huang, J. Seo, S. B. Willingham, A. M. Czyzewski, M. L. Gonzalgo, I. L. Weissman and A. E. Barron, *PLoS One*, 2014, **9**, e90397.
- S. A. Fowler and H. E. Blackwell, *Org. Biomol. Chem.*, 2009, **7**, 1508–1524.
- K. A. Brogden, A. J. DeLuca, J. Bland and S. Elliott, *Proc. Natl. Acad. Sci. U. S. A.*, 1996, **93**, 412–416.
- F. D. Silva, C. A. Rezende, D. C. P. Rossi, E. Esteves, F. H. Dyszy, S. Schreier, F. Gueiros-Filho, C. B. Campos, J. R. Pires and S. Daffre, *J. Biol. Chem.*, 2009, **284**, 34735–34746.
- G. N. Tew, R. W. Scott, M. L. Klein and W. F. Degrado, *Acc. Chem. Res.*, 2010, **43**, 30–39.
- M. S. Ganewatta and C. B. Tang, *Polymer*, 2015, **63**, A1–A29.
- W. Ren, W. R. Cheng, G. Wang and Y. Liu, *J. Polym. Sci., Part A: Polym. Chem.*, 2017, **55**, 632–639.
- L. Timofeeva and N. Kleshcheva, *Appl. Microbiol. Biotechnol.*, 2011, **89**, 475–492.
- A. Munoz-Bonilla and M. Fernandez-Garcia, *Prog. Polym. Sci.*, 2012, **37**, 281–339.
- H. Takahashi, E. F. Palermo, K. Yasuhara, G. A. Caputo and K. Kuroda, *Macromol. Biosci.*, 2013, **13**, 1285–1299.
- E. F. Palermo and K. Kuroda, *Appl. Microbiol. Biotechnol.*, 2010, **87**, 1605–1615.
- M. Hartlieb, E. G. L. Williams, A. Kuroki, S. Perrier and K. E. S. Locock, *Curr. Med. Chem.*, 2017, **24**, 2115–2140.
- J. Rodríguez-Hernández, in *Polymers against Microorganisms: On the Race to Efficient Antimicrobial Materials*, Springer International Publishing, Cham, 2017, pp. 39–69, DOI: 10.1007/978-3-319-47961-3\_3.
- A. C. Engler, N. Wiradharma, Z. Y. Ong, D. J. Coady, J. L. Hedrick and Y. Y. Yang, *Nano Today*, 2012, **7**, 201–222.
- M. R. E. Santos, A. C. Fonseca, P. V. Mendona, R. Branco, A. C. Serra, P. V. Morais and J. F. J. Coelho, *Materials*, 2016, **9**, 599.
- E.-R. Kenawy, S. D. Worley and R. Broughton, *Biomacromolecules*, 2007, **8**, 1359–1384.
- T. Ikeda, S. Tazuke and Y. Suzuki, *Makromol. Chem.*, 1984, **185**, 869–876.
- D. H. Liu and W. F. DeGrado, *J. Am. Chem. Soc.*, 2001, **123**, 7553–7559.
- M. A. Schmitt, B. Weisblum and S. H. Gellman, *J. Am. Chem. Soc.*, 2004, **126**, 6848–6849.
- Y. M. Song, Y. Park, S. S. Lim, S. T. Yang, E. R. Woo, I. S. Park, J. S. Lee, J. I. Kim, K. S. Hahm, Y. Kim and S. Y. Shin, *Biochemistry*, 2005, **44**, 12094–12106.



- 41 K. Matsuzaki, S. Yoneyama and K. Miyajima, *Biophys. J.*, 1997, **73**, 831–838.
- 42 L. Otvos and M. Cudic, in *Peptide Characterization and Application Protocols*, ed. G. B. Fields, Humana Press, Totowa, NJ, 2007, pp. 309–320, DOI: 10.1007/978-1-59745-430-8\_12.
- 43 K. Kuroda, G. A. Caputo and W. F. DeGrado, *Chem. – Eur. J.*, 2009, **15**, 1123–1133.
- 44 M. F. Ilker, K. Nusslein, G. N. Tew and E. B. Coughlin, *J. Am. Chem. Soc.*, 2004, **126**, 15870–15875.
- 45 B. P. Mowery, A. H. Lindner, B. Weisblum, S. S. Stahl and S. H. Gellman, *J. Am. Chem. Soc.*, 2009, **131**, 9735–9745.
- 46 B. P. Mowery, S. E. Lee, D. A. Kissounko, R. F. Epand, R. M. Epand, B. Weisblum, S. S. Stahl and S. H. Gellman, *J. Am. Chem. Soc.*, 2007, **129**, 15474–15476.
- 47 E. F. Palermo and K. Kuroda, *Biomacromolecules*, 2009, **10**, 1416–1428.
- 48 E. F. Palermo, D.-K. Lee, A. Ramamoorthy and K. Kuroda, *J. Phys. Chem. B*, 2011, **115**, 366–375.
- 49 M. A. Gelman, B. Weisblum, D. M. Lynn and S. H. Gellman, *Org. Lett.*, 2004, **6**, 557–560.
- 50 A. Punia, E. He, K. Lee, P. Banerjee and N. L. Yang, *Chem. Commun.*, 2014, **50**, 7071–7074.
- 51 A. Punia, P. R. Debata, P. Banerjee and N. L. Yang, *RSC Adv.*, 2015, **5**, 95300–95306.
- 52 M. F. Ilker, G. N. Tew and E. B. Coughlin, *Antiterrorism and Homeland Defense: Polymers and Materials*, 2007, vol. 980, pp. 175–197.
- 53 K. Lienkamp, A. E. Madkour and G. N. Tew, *Polymer Composites - Polyolefin Fractionation - Polymeric Peptidomimetics - Collagens*, 2013, vol. 251, pp. 141–172.
- 54 K. Lienkamp and G. N. Tew, *Chem. – Eur. J.*, 2009, **15**, 11784–11800.
- 55 K. Lienkamp, A. E. Madkour, A. Musante, C. F. Nelson, K. Nusslein and G. N. Tew, *J. Am. Chem. Soc.*, 2008, **130**, 9836–9843.
- 56 T. Eren, A. Som, J. R. Rennie, C. F. Nelson, Y. Urgina, K. Nusslein, E. B. Coughlin and G. N. Tew, *Macromol. Chem. Phys.*, 2008, **209**, 516–524.
- 57 Z. M. Al-Badri, A. Som, S. Lyon, C. F. Nelson, K. Nusslein and G. N. Tew, *Biomacromolecules*, 2008, **9**, 2805–2810.
- 58 H. Wang, X. F. Shi, D. F. Yu, J. Zhang, G. Yang, Y. X. Cui, K. J. Sun, J. B. Wang and H. K. Yan, *Langmuir*, 2015, **31**, 13469–13477.
- 59 V. W. L. Ng, J. P. K. Tan, J. Y. Leong, Z. X. Voo, J. L. Hedrick and Y. Y. Yang, *Macromolecules*, 2014, **47**, 1285–1291.
- 60 W. Chin, C. A. Yang, V. W. L. Ng, Y. Huang, J. C. Cheng, Y. W. Tong, D. J. Coady, W. M. Fan, J. L. Hedrick and Y. Y. Yang, *Macromolecules*, 2013, **46**, 8797–8807.
- 61 A. Nimmagadda, X. Liu, P. Teng, M. Su, Y. Q. Li, Q. Qiao, N. K. Khadka, X. T. Sung, J. J. Pan, H. Xu, Q. Li and J. F. Cai, *Biomacromolecules*, 2017, **18**, 87–95.
- 62 K. E. S. Locock, T. D. Michl, J. D. P. Valentin, K. Vasilev, J. D. Hayball, Y. Qu, A. Traven, H. J. Griesser, L. Meagher and M. Haeussler, *Biomacromolecules*, 2013, **14**, 4021–4031.
- 63 S. E. Exley, L. C. Paslay, G. S. Sahukhal, B. A. Abel, T. D. Brown, C. L. McCormick, S. Heinhorst, V. Koul, V. Choudhary, M. O. Elasri and S. E. Morgan, *Biomacromolecules*, 2015, **16**, 3845–3852.
- 64 A. Som, A. O. Tezgel, G. J. Gabriel and G. N. Tew, *Angew. Chem., Int. Ed.*, 2011, **50**, 6147–6150.
- 65 B. M. de Ronde, A. Birke and G. N. Tew, *Chem. – Eur. J.*, 2015, **21**, 3013–3019.
- 66 R. Tejero, D. Lopez, F. Lopez-Fabal, J. L. Gomez-Garces and M. Fernández-García, *Biomacromolecules*, 2015, **16**, 1844–1854.
- 67 Y. Chen, P. A. Wilbon, Y. P. Chen, J. H. Zhou, M. Nagarkatti, C. P. Wang, F. X. Chu, A. W. Decho and C. B. Tang, *RSC Adv.*, 2012, **2**, 10275–10282.
- 68 J. Y. Zhang, Y. P. Chen, K. P. Miller, M. S. Ganewatta, M. Bam, Y. Yan, M. Nagarkatti, A. W. Decho and C. B. Tang, *J. Am. Chem. Soc.*, 2014, **136**, 4873–4876.
- 69 V. Sambhy, B. R. Peterson and A. Sen, *Angew. Chem., Int. Ed.*, 2008, **47**, 1250–1254.
- 70 R. W. Williams, M. Zasloff and D. Covell, *Biophys. J.*, 1988, **53**, A631–A631.
- 71 M. Pillong, J. A. Hiss, P. Schneider, Y. C. Lin, G. Posselt, B. Pfeiffer, M. Blatter, A. T. Muller, S. Bachler, C. S. Neuhaus, P. S. Dittrich, K. H. Altmann, S. Wessler and G. Schneider, *Small*, 2017, **13**, 1701316.
- 72 A. Giangaspero, L. Sandri and A. Tossi, *Eur. J. Biochem.*, 2001, **268**, 5589–5600.
- 73 A. Tossi, C. Tarantino and D. Romeo, *Eur. J. Biochem.*, 1997, **250**, 549–558.
- 74 A. Som, S. Vemparala, I. Ivanov and G. N. Tew, *Biopolymers*, 2008, **90**, 83–93.
- 75 G. J. Gabriel, J. A. Maegerlein, C. E. Nelson, J. M. Dabkowski, T. Eren, K. Nusslein and G. N. Tew, *Chem. – Eur. J.*, 2009, **15**, 433–439.
- 76 H. Takahashi, G. A. Caputo, S. Vemparala and K. Kuroda, *Bioconjugate Chem.*, 2017, **28**, 1340–1350.
- 77 K. Lienkamp, K. N. Kumar, A. Som, K. Nusslein and G. N. Tew, *Chem. – Eur. J.*, 2009, **15**, 11710–11714.
- 78 T. D. Michl, K. E. S. Locock, N. E. Stevens, J. D. Hayball, K. Vasilev, A. Postma, Y. Qu, A. Traven, M. Haeussler, L. Meagher and H. J. Griesser, *Polym. Chem.*, 2014, **5**, 5813–5822.
- 79 J. L. Grace, J. X. Huang, S. E. Cheah, N. P. Truong, M. A. Cooper, J. Li, T. P. Davis, J. F. Quinn, T. Velkov and M. R. Whittaker, *RSC Adv.*, 2016, **6**, 15469–15477.
- 80 J. L. Grace, A. G. Elliott, J. X. Huang, E. K. Schneider, N. P. Truong, M. A. Cooper, J. Li, T. P. Davis, J. F. Quinn, T. Velkov and M. R. Whittaker, *J. Mater. Chem. B*, 2017, **5**, 531–536.
- 81 D. J. Coady, Z. Y. Ong, P. S. Lee, S. Venkataraman, W. Chin, A. C. Engler, Y. Y. Yang and J. L. Hedrick, *Adv. Healthcare Mater.*, 2014, **3**, 882–889.
- 82 K. Hu, N. W. Schmidt, R. Zhu, Y. Jiang, G. H. Lai, G. Wei, E. F. Palermo, K. Kuroda, G. C. L. Wong and L. Yang, *Macromolecules*, 2013, **46**, 1908–1915.



- 83 B. C. Allison, B. M. Applegate and J. P. Youngblood, *Biomacromolecules*, 2007, **8**, 2995–2999.
- 84 S. Colak, C. F. Nelson, K. Nusslein and G. N. Tew, *Biomacromolecules*, 2009, **10**, 353–359.
- 85 A. Punia, A. Mancuso, P. Banerjee and N. L. Yang, *ACS Macro Lett.*, 2015, **4**, 426–430.
- 86 S. Chakraborty, R. H. Liu, Z. Hayouka, X. Y. Chen, J. Ehrhardt, Q. Lu, E. Burke, Y. Q. Yan, B. Weisblum, G. C. L. Wong, K. S. Masters and S. H. Gellman, *J. Am. Chem. Soc.*, 2014, **136**, 14530–14535.
- 87 X. Yang, K. Hu, G. T. Hu, D. Y. Shi, Y. J. Jiang, L. W. Hui, R. Zhu, Y. T. Xie and L. H. Yang, *Biomacromolecules*, 2014, **15**, 3267–3277.
- 88 A. Punia, K. Lee, E. He, S. Mukherjee, A. Mancuso, P. Banerjee and N. L. Yang, *Int. J. Mol. Sci.*, 2015, **16**, 23867–23880.
- 89 M. Alvarez-Paino, A. Munoz-Bonilla, F. Lopez-Fabal, J. L. Gomez-Garces, J. P. A. Heuts and M. Fernandez-Garcia, *Biomacromolecules*, 2015, **16**, 295–303.
- 90 E. H. H. Wong, M. M. Khin, V. Ravikumar, Z. Y. Si, S. A. Rice and M. B. Chan-Park, *Biomacromolecules*, 2016, **17**, 1170–1178.
- 91 T. K. Nguyen, S. J. Lam, K. K. K. Ho, N. Kumar, G. G. Qiao, S. Egan, C. Boyer and E. H. H. Wong, *ACS Infect. Dis.*, 2017, **3**, 237–248.
- 92 R. Namivandi-Zangeneh, R. J. Kwan, T.-K. Nguyen, J. Yeow, F. L. Byrne, S. H. Oehlers, E. H. H. Wong and C. Boyer, *Polym. Chem.*, 2018, DOI: 10.1039/C7PY01069A, advance article.
- 93 S. J. Lam, N. M. O'Brien-Simpson, N. Pantarat, A. Sulistio, E. H. H. Wong, Y. Y. Chen, J. C. Lenzo, J. A. Holden, A. Blencowe, E. C. Reynolds and G. G. Qiao, *Nat. Microbiol.*, 2016, **1**, 16162.
- 94 Y. J. Jiang, W. Zheng, L. J. Kuang, H. R. Ma and H. J. Liang, *ACS Infect. Dis.*, 2017, **3**, 676–687.
- 95 E. Strandberg, S. Morein, P. C. A. van der Wel, R. E. Koeppe and J. A. Killian, *Biophys. J.*, 2001, **80**, 541a–542a.
- 96 E. Strandberg and J. A. Killian, *FEBS Lett.*, 2003, **544**, 69–73.
- 97 E. F. Palermo, S. Vemparala and K. Kuroda, *Biomacromolecules*, 2012, **13**, 1632–1641.
- 98 F. Nederberg, Y. Zhang, J. P. K. Tan, K. J. Xu, H. Y. Wang, C. Yang, S. J. Gao, X. D. Guo, K. Fukushima, L. J. Li, J. L. Hedrick and Y. Y. Yang, *Nat. Chem.*, 2011, **3**, 409–414.
- 99 M. Mizutani, E. F. Palermo, L. M. Thoma, K. Satoh, M. Kamigaito and K. Kuroda, *Biomacromolecules*, 2012, **13**, 1554–1563.
- 100 E. A. Chamsaz, S. Mankoci, H. A. Barton and A. Joy, *ACS Appl. Mater. Interfaces*, 2017, **9**, 6704–6711.
- 101 J. A. Grapski and S. L. Cooper, *Biomaterials*, 2001, **22**, 2239–2246.
- 102 S. Mankoci, R. L. Kaiser, N. Sahai, H. A. Barton and A. Joy, *ACS Biomater. Sci. Eng.*, 2017, **3**, 2588–2597.
- 103 D. N. Amato, D. V. Amato, O. V. Mavrodi, W. B. Martin, S. N. Swilley, K. H. Parsons, D. V. Mavrodi and D. L. Patton, *ACS Macro Lett.*, 2017, **6**, 171–175.
- 104 M. Vert, *Biomacromolecules*, 2005, **6**, 538–546.
- 105 M. A. Dewit and E. R. Gillies, *J. Am. Chem. Soc.*, 2009, **131**, 18327–18334.
- 106 A. Sagi, R. Weinstain, N. Karton and D. Shabat, *J. Am. Chem. Soc.*, 2008, **130**, 5434–5435.
- 107 A. M. DiLauro, J. S. Robbins and S. T. Phillips, *Macromolecules*, 2013, **46**, 2963–2968.
- 108 S. T. Phillips and A. M. DiLauro, *ACS Macro Lett.*, 2014, **3**, 298–304.
- 109 C. E. Diesendruck, G. I. Peterson, H. J. Kulik, J. A. Kaitz, B. D. Mar, P. A. May, S. R. White, T. J. Martinez, A. J. Boydston and J. S. Moore, *Nat. Chem.*, 2014, **6**, 624–629.
- 110 G. I. Peterson, M. B. Larsen and A. J. Boydston, *Macromolecules*, 2012, **45**, 7317–7328.
- 111 M. A. Dewit, A. Beaton and E. R. Gillies, *J. Polym. Sci., Part A: Polym. Chem.*, 2010, **48**, 3977–3985.
- 112 B. Fan, J. F. Trant, A. D. Wong and E. R. Gillies, *J. Am. Chem. Soc.*, 2014, **136**, 10116–10123.
- 113 A. D. Wong, M. A. DeWit and E. R. Gillies, *Adv. Drug Delivery Rev.*, 2012, **64**, 1031–1045.
- 114 K. Yeung, H. Kim, H. Mohapatra and S. T. Phillips, *J. Am. Chem. Soc.*, 2015, **137**, 5324–5327.
- 115 A. P. Esser-Kahn, N. R. Sottos, S. R. White and J. S. Moore, *J. Am. Chem. Soc.*, 2010, **132**, 10266–10268.
- 116 M. Shamis, H. N. Lode and D. Shabat, *J. Am. Chem. Soc.*, 2004, **126**, 1726–1731.
- 117 H. Wang, Q. Huang, H. Chang, J. R. Xiao and Y. Y. Cheng, *Biomater. Sci.*, 2016, **4**, 375–390.
- 118 Y. Xie, T. Murray-Stewart, Y. Z. Wang, F. Yua, J. Li, L. J. Marton, R. A. Casero and D. Oupicky, *J. Controlled Release*, 2017, **246**, 110–119.
- 119 A. M. DiLauro, G. G. Lewis and S. T. Phillips, *Angew. Chem., Int. Ed.*, 2015, **54**, 6200–6205.
- 120 H. Zhang, K. Yeung, J. S. Robbins, R. A. Pavlick, M. Wu, R. Liu, A. Sen and S. T. Phillips, *Angew. Chem., Int. Ed.*, 2012, **51**, 2400–2404.
- 121 A. P. Esser-Kahn, S. A. Odom, N. R. Sottos, S. R. White and J. S. Moore, *Macromolecules*, 2011, **44**, 5539–5553.
- 122 C. Ergene and E. F. Palermo, *Biomacromolecules*, 2017, **18**, 3400–3409.
- 123 M. G. Olah, J. S. Robbins, M. S. Baker and S. T. Phillips, *Macromolecules*, 2013, **46**, 5924–5928.
- 124 J. F. Lutz, *Polym. Chem.*, 2010, **1**, 55–62.
- 125 K. Hilpert, M. R. Elliott, R. Volkmer-Engert, P. Henklein, O. Donini, Q. Zhou, D. F. H. Winkler and R. E. W. Hancock, *Chem. Biol.*, 2006, **13**, 1101–1107.
- 126 Y. Oda, S. Kanaoka, T. Sato, S. Aoshima and K. Kuroda, *Biomacromolecules*, 2011, **12**, 3581–3591.
- 127 A. R. Song, S. G. Walker, K. A. Parker and N. S. Sampson, *ACS Chem. Biol.*, 2011, **6**, 590–599.
- 128 D. S. S. M. Uppu, M. Bhowmik, S. Samaddar and J. Haldar, *Chem. Commun.*, 2016, **52**, 4644–4647.
- 129 A. Kuroki, P. Sangwan, Y. Qu, R. Peltier, C. Sanchez-Cano, J. Moat, C. G. Dowson, E. G. L. Williams, K. E. S. Locock, M. Hartlieb and S. Perrier, *ACS Appl. Mater. Interfaces*, 2017, **9**(46), 40117–40126.



- 130 L. M. Thoma, B. R. Boles and K. Kuroda, *Biomacromolecules*, 2014, **15**, 2933–2943.
- 131 D. S. S. M. Uppu, S. Samaddar, J. Hoque, M. M. Konai, P. Krishnamoorthy, B. R. Shome and J. Haldar, *Biomacromolecules*, 2016, **17**, 3094–3102.
- 132 M. M. Konai and J. Haldar, *Bioconjugate Chem.*, 2017, **28**, 1194–1204.
- 133 Q. Gao, M. Yu, Y. J. Su, M. H. Xie, X. Zhao, P. Li and P. X. Ma, *Acta Biomater.*, 2017, **51**, 112–124.
- 134 J. C. Cheng, W. Chin, H. H. Dong, L. Xu, G. S. Zhong, Y. Huang, L. J. Li, K. J. Xu, M. Wu, J. L. Hedrick, Y. Y. Yang and W. M. Fan, *Adv. Healthcare Mater.*, 2015, **4**, 2128–2136.
- 135 H. G. Sahl, *Microbiol. Sci.*, 1985, **2**, 212–217.
- 136 K. Matsuzaki, *Biochim. Biophys. Acta, Biomembr.*, 1999, **1462**, 1–10.
- 137 L. Yang, T. M. Weiss, R. I. Lehrer and H. W. Huang, *Biophys. J.*, 2000, **79**, 2002–2009.
- 138 K. Matsuzaki, O. Murase and K. Miyajima, *Biochemistry*, 1995, **34**, 12553–12559.
- 139 G. Bocchinfuso, A. Palleschi, B. Orioni, G. Grande, F. Formaggio, C. Toniolo, Y. Park, K. S. Hahm and L. Stella, *J. Pept. Sci.*, 2009, **15**, 550–558.
- 140 Y. Shai, *Biochim. Biophys. Acta, Biomembr.*, 2002, **66**, 236–248.
- 141 D. Sengupta, H. Leontiadou, A. E. Mark and S. J. Marrink, *Biochim. Biophys. Acta, Biomembr.*, 2008, **1778**, 2308–2317.
- 142 E. Gazit, I. R. Miller, P. C. Biggin, M. S. P. Sansom and Y. Shai, *J. Mol. Biol.*, 1996, **258**, 860–870.
- 143 A. J. Krauson, J. He and W. C. Wimley, *Biochim. Biophys. Acta, Biomembr.*, 2012, **1818**, 1625–1632.
- 144 K. W. Bayles, *Nat. Rev. Microbiol.*, 2014, **12**, 63–69.
- 145 L. P. Ding, E. Y. Chi, S. Chemburu, E. Ji, K. S. Schanze, G. P. Lopez and D. G. Whitten, *Langmuir*, 2009, **25**, 13742–13751.
- 146 K. Matsuzaki, K. Sugishita, N. Fujii and K. Miyajima, *Biochemistry*, 1995, **34**, 3423–3429.
- 147 S. G. Hovakeemian, R. H. Liu, S. H. Gellman and H. Heerklotz, *Soft Matter*, 2015, **11**, 6840–6851.
- 148 J. R. Lai, R. F. Epand, B. Weisblum, R. M. Epand and S. H. Gellman, *Biochemistry*, 2006, **45**, 15718–15730.
- 149 K. Matsuzaki, K. Sugishita, N. Ishibe, M. Ueha, S. Nakata, K. Miyajima and R. M. Epand, *Biochemistry*, 1998, **37**, 11856–11863.
- 150 Y. H. Li, L. M. Chan, L. Tyer, R. T. Moody, C. M. Himel and D. M. Hercules, *J. Am. Chem. Soc.*, 1975, **97**, 3118–3126.
- 151 E. F. Palermo, D. K. Lee, A. Ramamoorthy and K. Kuroda, *J. Phys. Chem. B*, 2011, **115**, 366–375.
- 152 Y. Tamba and M. Yamazaki, *Biochemistry*, 2005, **44**, 15823–15833.
- 153 J. M. Alam, T. Kobayashi and M. Yamazaki, *Biochemistry*, 2012, **51**, 5160–5172.
- 154 E. E. Ambroggio, F. Separovic, J. H. Bowie, G. D. Fidelio and L. A. Bagatolli, *Biophys. J.*, 2005, **89**, 1874–1881.
- 155 T. M. Domingues, K. A. Riske and A. Miranda, *Langmuir*, 2010, **26**, 11077–11084.
- 156 L. H. Yang, V. D. Gordon, A. Mishra, A. Sorn, K. R. Purdy, M. A. Davis, G. N. Tew and G. C. L. Wong, *J. Am. Chem. Soc.*, 2007, **129**, 12141–12147.
- 157 K. Hu, N. W. Schmidt, R. Zhu, Y. J. Jiang, G. H. Lai, G. Wei, E. F. Palermo, K. Kuroda, G. C. L. Wong and L. H. Yang, *Macromolecules*, 2013, **46**, 1908–1915.
- 158 R. F. Epand, G. S. Wang, B. Berno and R. M. Epand, *Antimicrob. Agents Chemother.*, 2009, **53**, 3705–3714.
- 159 R. F. Epand, W. L. Maloy, A. Ramamoorthy and R. M. Epand, *Biochemistry*, 2010, **49**, 4076–4084.
- 160 R. M. Epand, S. Rotem, A. Mor, B. Berno and R. F. Epand, *J. Am. Chem. Soc.*, 2008, **130**, 14346–14352.
- 161 R. F. Epand, B. P. Mowery, S. E. Lee, S. S. Stahl, R. I. Lehrer, S. H. Gellman and R. M. Epand, *J. Mol. Biol.*, 2008, **379**, 38–50.
- 162 C. W. Avery, A. Som, Y. J. Xu, G. N. Tew and Z. Chen, *Anal. Chem.*, 2009, **81**, 8365–8372.
- 163 C. W. Avery, E. F. Palermo, A. McLaughlin, K. Kuroda and Z. Chen, *Anal. Chem.*, 2011, **83**, 1342–1349.
- 164 X. Y. Chen, J. Wang, A. P. Boughton, C. B. Kristalyn and Z. Chen, *J. Am. Chem. Soc.*, 2007, **129**, 1420–1427.
- 165 A. Ramamoorthy, *Solid State Nucl. Magn. Reson.*, 2009, **35**, 201–207.
- 166 Y. C. Su, W. F. DeGrado and M. Hong, *J. Am. Chem. Soc.*, 2010, **132**, 9197–9205.
- 167 I. Ivanov, S. Vemparala, V. Pophristic, K. Kuroda, W. F. DeGrado, J. A. McCammon and M. L. Klein, *J. Am. Chem. Soc.*, 2006, **128**, 1778–1779.
- 168 E. Strandberg, S. Morein, D. T. S. Rijkers, R. M. J. Liskamp, P. C. A. van der Wel and J. A. Killian, *Biochemistry*, 2002, **41**, 7190–7198.
- 169 V. K. Mishra, M. N. Palgunachari, J. P. Segrest and G. M. Anantharamaiah, *J. Biol. Chem.*, 1994, **269**, 7185–7191.
- 170 U. Baul and S. Vemparala, *Soft Matter*, 2017, **13**, 7665–7676.
- 171 U. Baul, K. Kuroda and S. Vemparala, *J. Chem. Phys.*, 2014, **141**, 084902.
- 172 M. Benincasa, S. Pacor, R. Gennaro and M. Scocchi, *Antimicrob. Agents Chemother.*, 2009, **53**, 3501–3504.
- 173 C. S. Alves, M. N. Melo, H. G. Franquelim, R. Ferre, M. Planas, L. Feliu, E. Bardaji, W. Kowalczyk, D. Andreu, N. C. Santos, M. X. Fernandes and M. A. R. B. Castanho, *J. Biol. Chem.*, 2010, **285**, 27536–27544.
- 174 K. S. Huang, C. H. Yang, S. L. Huang, C. Y. Chen, Y. Y. Lu and Y. S. Lin, *Int. J. Mol. Sci.*, 2016, **17**.
- 175 C. D. Fjell, J. A. Hiss, R. E. W. Hancock and G. Schneider, *Nat. Rev. Drug Discovery*, 2012, **11**, 37–51.

