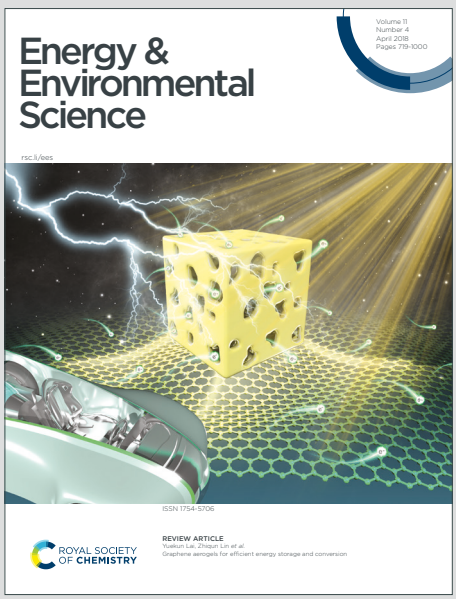
Energy & Environmental Science

Accepted Manuscript

This article can be cited before page numbers have been issued, to do this please use: Q. Chen, W. Zhu, Y. Chen, H. An, S. Yang, Y. Wang, Y. Ji, G. Su, R. Wang, J. E. Huang, J. Song, J. Kim, W. Ni, C. Musgrave, K. Xie and E. H. Sargent, *Energy Environ. Sci.*, 2026, DOI: 10.1039/D5EE04672F.



ROYAL SOCIETY OF CHEMISTRY

rsc.li/ees

View Article Online
DOI: 10.1039/D5EE04672F

Judd A. and Marjorie Weinberg College of Arts and Sciences
Northwestern University

Department of Chemistry
2145 Sheridan Road
Evanston, Illinois 60208-3113

Ted Sargent
Professor of Chemistry

ted.sargent@northwestern.edu

Broader Context

Electrochemical conversion of CO_2-derived CO into liquid fuels like ethanol offers a promising route to decarbonize heavy industry and achieve carbon neutrality. However, existing systems face a critical trade-off: high alkalinity boosts liquid fuel production but accelerates corrosion, limiting device stability and practicality. This work addresses this challenge by designing an asymmetric bipolar membrane electrode assembly (BPMEA) that locally generates an ultra-high pH environment (pH $\approx$ 15) at the catalyst surface while maintaining benign bulk conditions. We demonstrate that this membrane architecture achieves 93% selectivity for liquid products, with a 70:1 ethanol-to-ethylene ratio, and operates stably for 28 hours (versus <30 minutes in conventional systems). Crucially, it reduces liquid product crossover by 12x and achieves unprecedented ethanol concentrations (23 wt%) on the cathode. By revealing how localized hydroxide superconcentration steers reaction pathways toward ethanol, this work provides a blueprint for stable, membrane-engineered electrolyzers. This advance bridges a key gap toward scalable production of carbon-neutral liquid fuels from CO_2.

View Article Online
DOI: 10.1039/D5EE04672F

High-Asymmetry Bipolar Membrane Electrode Assemblies Generate a Superconcentration of Cations and Hydroxide at a Catalyst Surface

Authors: Qiu-Cheng Chen[1]†, Wenjin Zhu[1]†, Yiqing Chen[1]†, Hongmin An[1, 3], Shuang Yang[1], Yong Wang[1], Yali Ji[1], Guangcan Su[1], Rui Wang[1], Jianan Erick Huang[1], Ji-Yoon Song[1], Jaerim Kim[1], Weiyan Ni[1], Charles Musgrave[1], Ke Xie[1]*, Edward H. Sargent[1, 2]*

Affiliations:

[1]Department of Chemistry, Northwestern University, Evanston, IL, USA

[2]Department of Electrical and Computer Engineering, Northwestern University, Evanston, IL, USA

[3]Department of Chemical and Biomolecular Engineering, Korea Advanced Institute of Science and Technology, Daejeon, Republic of Korea

†These authors contributed equally

*Corresponding author. Email: ted.sargent@northwestern.edu; ke-xie@northwestern.edu

Abstract

In electrochemical CO reduction reactions, a highly alkaline pH is typically desired to promote multicarbon liquid products and suppress hydrogen evolution, considerations that prioritize pH ≥ 14 (e.g. 1 M KOH). However, bulk electrolytes with pH exceeding 14 are prone to produce corrosion of catalyst and electrolyzer. Here we find that an engineered class of bipolar membrane assemblies (BPMEAs) achieves a superconcentration of local metal hydroxides, and generates a product slate consistent with local electrolyte pH = 15. We report that, in a cathode:anion exchange layer (AEL):cation exchange layer (CEL):anode architecture, a high thickness ratio of CEL:AEL generates a high local pH at the cathode, this achieved by blocking the transport of hydroxide ions, generated on the cathode, over to the anode side. This enables production of C_{2+} liquids at a total Faradaic efficiency of 93%, with an ethanol:ethylene productivity ratio of 70:1. Compared to anion-exchange membrane assemblies (AEMEAs) operating at the same 100 mA cm^{-2} current density for similar product selectivity, these BPMEA systems exhibit 28 hours stable operation (compared to <30 minutes in AEMEA), and a 12x lower rate of liquid product crossover, enabling us to report a liquid product concentration of 23 wt% on the cathode. Operando Raman spectroscopy shows that the optimal BPM enhances coverage, on the cathode catalyst, of surface-bound hydroxyl species, ~ 5x higher than AEM systems, simultaneous with maximizing the surface CO population. Mechanistic studies indicate that surface OH promotes hydroxylation of the CCH intermediate, steering the reaction pathway toward ethanol instead of ethylene, leading to the strong preference towards liquid production.

View Article Online
DOI: 10.1039/D5EE04672F

Introduction

The CO electroreduction reaction (CORR) is of interest in making multi-carbon products, and CO is generated efficiently from CO_2 in solid-oxide electrolyzers.[1-4] Compared with gaseous CO_2RR products, liquid products enable convenient transport from the point of production to that of further processing or utilization.[5] Since ethylene and ethanol share key intermediates in reaction mechanisms, significant efforts have focused on enhancing ethanol and liquid-product selectivity by modulating these competing pathways via catalyst engineering[3, 6-12] as well as through the design of membrane,[13-17] ionomer,[18-19] and electrolyte.[20-29]

Alkalinity is known to facilitate C-C coupling and suppress HER in favour of CORR,[1, 30-31] and has been shown to enhance selectivity to acetate[1, 32-33]. Recent studies also suggested that the adsorbed OH[7, 34-36] is a key species to turn on the $CO_{(2)}$-to-ethanol pathway, entailing that ethanol selectivity may also be increased by manipulating alkalinity.[37-38] In these studies, the coverage of adsorbed OH and CO was tuned by catalyst modification and pulsed oxidation of a copper catalyst surface. Notably, the dependence of both ethanol production and the ethanol-to-ethylene ratio on bulk pH and surface OH/CO coverage have been proposed to be nonmonotonic, indicating a complex ethanol/ethylene production mechanism.[34-35] In principle, increasing electrolyte alkalinity can directly regulate OH coverage. Indeed, we found that gradually increasing anolyte concentration steers CORR toward ethanol production (Figure S1). However, such a high alkalinity can destabilize oxygen evolution reaction (OER) catalysts[39-40] and induce corrosion[41-42] (Figure S2).

In parallel to catalyst design, advancing CORR electrocatalysis relies increasingly on zero-gap membrane electrode assembly (MEA) electrolyzers to regulate reaction conditions.[43-44] In these systems, the cathode and anode are disposed on either side of a polymeric ion exchange membrane in a zero-gap configuration, with gas supplied to the cathodic gas diffusion electrode (GDE), and electrolyte circulated on the anode (Figure 1a).[13-14, 16, 45-49] The choice of membrane will affect the pH and ions on the catalyst layers (Figure 1b), affecting thereby the reaction pathway and thus the product distribution. The impact of membrane/assembly design on pH and ion egress has been studied and exploited in water electrolyzers.[50-54]

We focused herein on how membrane design impacts surface coverage of OH and CO when the composite membrane is rendered selective for the transport of specific ions[49, 54-56], our goal to favor liquid production.

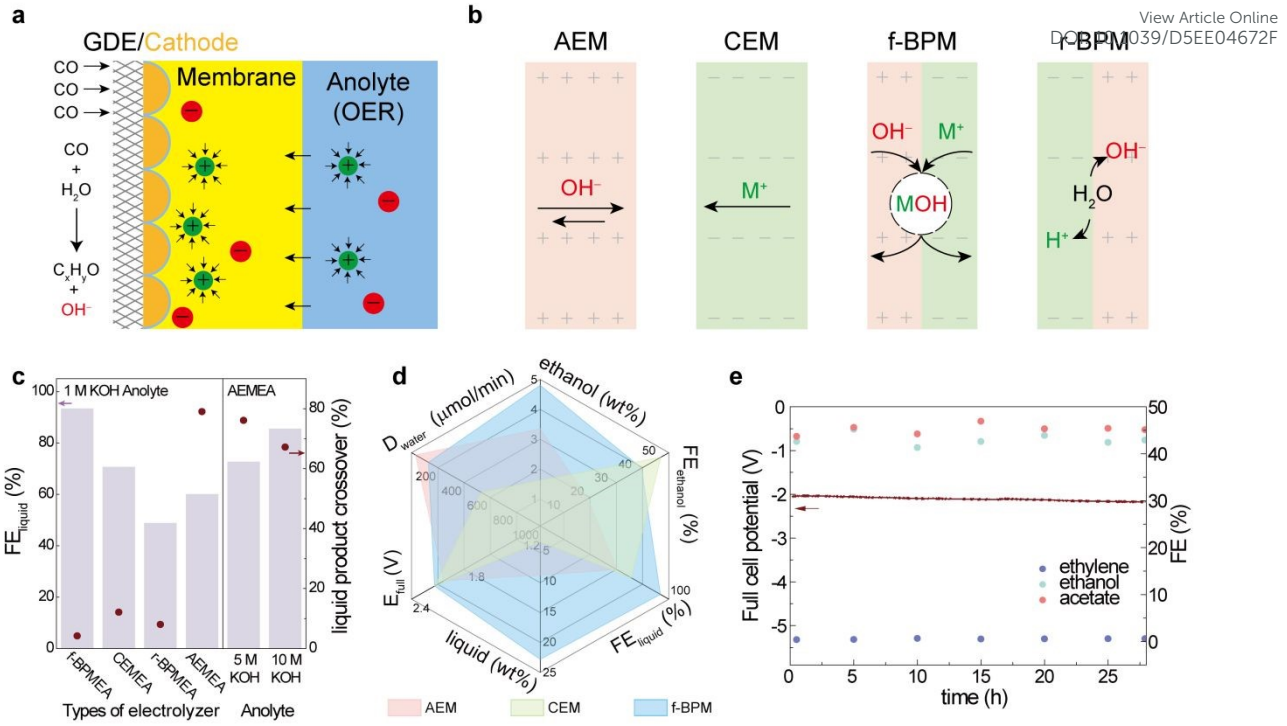


Figure 1. MEA-CO electrolyzer configuration and membrane-dependent CORR performance. a. Schematic illustrations of the interface structure in the MEA. **b.** Transport of ionic species (M^+ = Li^+, Na^+, K^+, Cs^+ etc.) in different membrane configurations: anion exchange membrane (AEM), cation exchange membrane (CEM), forward-bias bipolar membrane (f-BPM) and reverse-bias bipolar membrane (r-BPM). **c.** Faradaic efficiency and liquid-product crossover of CORR liquid products in MEA with different membrane or anolyte. (**d**) Key performance metrics (water migration rate, D_{water}; FEs and concentration of ethanol and total liquid product; full cell voltages, E_{full}) for AEM, CEM, and f-BPM cells using 1 M KOH anolyte; (**e**) Cell voltage and Faradaic efficiencies for CO reduction in the optimized f-BPMEA system during a 28-hour operation.

Results

CORR selectivity under different membrane configurations

We began by investigating CORR reactivity in a suite of candidate membrane configurations: AEM, CEM, forward-bias BPM and reverse-bias BPM; all in a zero-gap cell (Figure 1a & S3) at 100 mA cm^{-2} current density. In 1 M KOH anolyte, the AEM cell (AEMEA) shows FE_{liquid} of 60%, and the CEM and r-BPM cells (CEMEA and r-BPMEA) show 72% and 49%, respectively. In contrast, the f-BPM cell (f-BPMEA) achieves FE_{liquid} of 93% and correspondingly low gas production (Figure 1c & S4).

Compared to AEM, the cation exchange layer (CEL) in f-BPM serves as a blocker for OH^- generated during CORR (Figure 1b, AEM vs. f-BPM), while transferring K^+ from the anolyte, instead of H^+ in r-BPM systems (Figure 1b, r-BPM vs. f-BPM). The result is an increased KOH concentration on the cathode surface, known to promote CORR to multi-carbon products.[1, 20-21, 23] The product distribution is similar in an AEMEA in which a higher anolyte concentration,

View Article Online
DOI: 10.1039/D5EE04672F

10 M KOH, is employed; however, these systems suffer from instability (Figure S2) and a >12x higher liquid product crossover rate compared to f-BPMEA (Figure 1c), the result of electroosmotic drag effect in AEMEA. The f-BPMEA achieved a 2.5x higher liquid product concentration on the cathode compared to AEMEA (23 wt% vs. 9 wt%, Figure 1d, liquid wt%) and it operated at a similar full-cell voltage (Figure 1d, E_{full}).

In CEMEA, K^+ migrates from anolyte to cathode, with electroosmotic drag leading to the transport of water molecules (one study[57] quantified 27 H_2O per K^+). This ion current inhibits the undesired transport of liquid product to the anolyte, but the co-transport of H_2O from anode to cathode causes unacceptably high product dilution on the cathode. We observed that f-BPM hinders anode-to-cathode water crossover (Figure 1d, D_{water}, rate of water crossover), and, when one compares to CEM, the cathode-side liquid product concentration is 6x higher (Figure 1d, liquid wt%). The AEL of f-BPM provides cathode-to-anode OH^- egress, dragging 4 – 7 H_2O per OH^-.[57] This anion/water flow offsets the cation/water flow. We ran the f-BPMEA continuously for 28 hours, and found that it maintained high selectivity of ethanol and acetate over ethylene at 100 mA cm^{-2} (Figure 1e). The corresponding energy efficiencies (EE) for ethanol, and for total liquid products, were 22% and 40%, respectively. The concentrated ethanol output from the f-BPMEA enables a distillation cost that is less than half of that for the CEMEA (Supplementary Note 5). While smaller cations (Li^+, Na^+) are anticipated to transport less water owing to their lower hydration numbers,[58] the use of LiOH and NaOH as anolytes was found to shift product selectivity toward gas-phase products (Figure S17).

We then studied the impact of the thickness ratio of CEL to AEL on the CORR selectivity. In the AEMEA, n-propanol (31% FE) and ethylene (30% FE) are the main products, with 21% of $FE_{ethanol}$ (Figure 2a). We processed CELs of different thicknesses atop a prefabricated anion exchange layer (AEL) in order to produce BPMs, and placed these in the MEA in the forward bias configuration. Increasing the thickness of CEL from 2 to 25 um (0.04 to 0.5 CEL:AEL thickness ratio) leads to diminished FE_{HER}, $FE_{ethylene}$ and $FE_{n\text{-}propanol}$, and progressively increasing selectivity toward ethanol and acetate (Figure 2a). Further increasing the thickness ratio of CEL:AEL to 0.5 suppresses the FE of hydrogen and ethylene evolution reactions to below 1%. Interestingly, we then saw a rise in both HER and ethylene production, and decreased ethanol and acetate selectivity, when we increased the CEL:AEL thickness ratio beyond 0.5 (Figure 2a). Assuming the CEL plays a role in keeping the OH^- at the cathode side, a thicker CEL decreases the transport rate of OH^- from the cathode surface to the anolyte, increasing the local pH of the catalyst layer.

To explore the relationship between the membranes and the interfacial environment, we studied ion transport in the MEA. We track K^+, instead of OH^-, since the latter could originate from anolyte crossover or cathodic reactions. At the cathode side after CORR, under 100 mA cm^{-2} galvanostatic electrolysis, we estimated the rate of K^+ transport (D_{K+}) from anolyte to cathode by collecting the totality of liquids and gases exiting the cathode chamber and the gas outlet,

View Article Online
DOI: 10.1039/D5EE04672F

and evaluated composition using inductively coupled plasma (ICP) spectroscopy (Supplementary note 1). A rate of 0.5 μmol/min for D_{K+} was observed in the AEMEA system. When we introduced a CEL onto the AEM to form a f-BPM, we saw evidence of an increased D_{K+} value, with only 2 μm of CEL leading to 12x higher D_{K+} than bare AEM (Figure 2b). Thicker CEL@f-BPM further accelerates D_{K+}, reaching 24 μmol/min D_{K+} with a 1-to-1 CEL:AEL thickness ratio. Since (1) the ion diffusion constant (c_i), electromigration (ϕ) and reaction rate are similar under identical applied current density and temperature in AEL, and (2) CEL transport of OH^- is much slower than of K^+, the OH^- generated during the COR began to accumulate between the cathode and the CEL, leading to K^+ transport across the CEL from anode to cathode through ion-correlations (μ_i) that achieve electroneutrality at the cathode surface.[59] When we changed the anolyte from 1 M KOH to neutral 0.5 M K_2SO_4, we saw similar D_{K+} and CORR product distribution, from which we surmise that the cathode surface environment is independent of the anolyte (Figure S5).

We propose therefore this picture of the f-BPM system: the majority of OH^- generated from the cathodic reaction accumulates between the AEL and CEL, forming superconcentrated KOH at the interface. KOH diffuses through the AEL and causes the high alkalinity of the cathodic local environmental (Figure 2c).[16, 60] COMSOL simulation of the ion distribution within f-BPMs vs. thickness ratios also supports the higher CEL:AEL ratio leading to higher pH and $[K^+]$ near the cathode catalyst layer (Figure 2d, supplementary note 2).

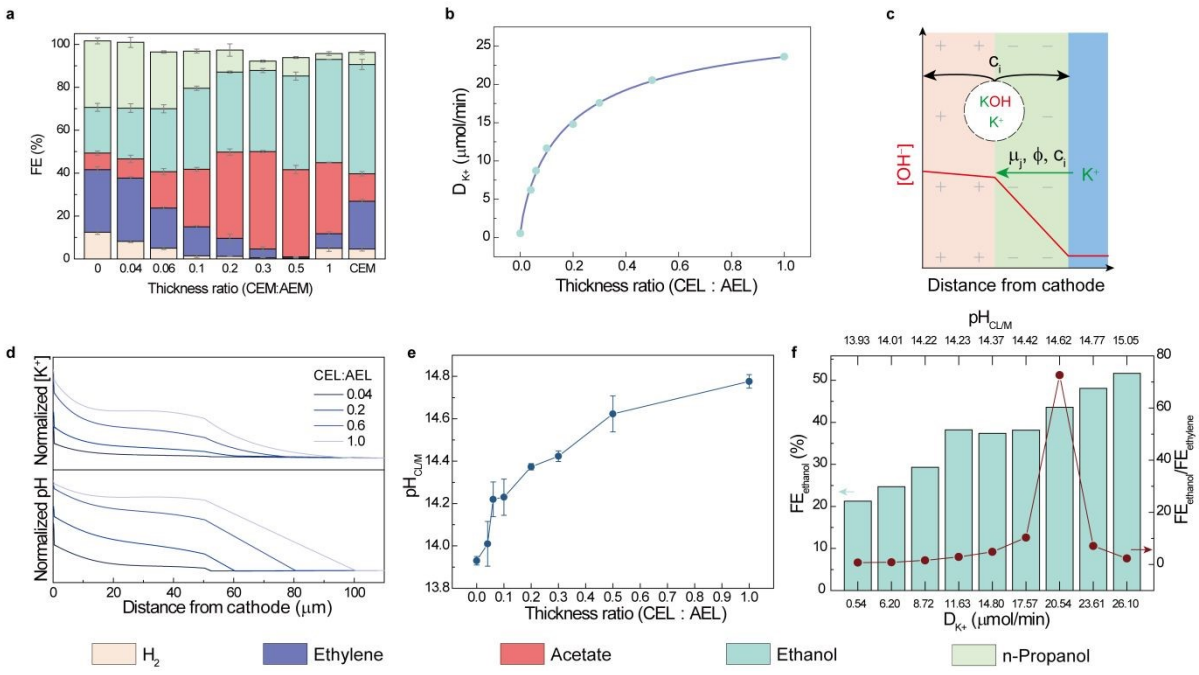


Figure 2. Characteristics of f-BPM for CORR in MEA systems. a Faradaic efficiency profiles for CO reduction on Cu catalyst using AEM, CEM, and f-BPM with varying CEL-to-AEL thickness ratios at -100 mA cm^{-2}. **b.** Potassium transport rate (D_{K+}) in membranes with different CEL-to-AEL ratios; a ratio of 0 corresponds to pristine AEM. **c.** Schematic of species transport mechanisms in the MEA with an f-BPM. **d** COMSOL-simulated species

View Article Online
DOI: 10.1039/D5EE04672F

concentration profiles from cathode to anolyte for varying CEL-to-AEL thickness ratios. **e.** Measured interfacial pH (cathode/membrane) for different CEL-to-AEL thickness ratios. **f.** $FE_{ethanol}$ and ethanol-to-ethylene selectivity as functions of D_{K+} and interfacial pH ($pH_{CL/M}$) for tested membrane configurations.

To quantify the pH in the interface between the catalyst layer (CL) and membrane ($pH_{CL/M}$), we carried out CORR in a modified MEA cell equipped with a needle sensor that provides a metal/metal oxide redox couple (Supplementary Note 3, Figure S6-S7). The observed $pH_{CL/M}$ value was 13.93 in the AEM system, and this increases rapidly to 14.37 at CEL:AEL = 0.2 and continues to 14.77 at 1:1 ratio (Figure 2e & Figure S7). This is consistent with the COMSOL study, in which larger CEM:AEM ratio led to a higher pH in the CL-to-AEM interface than that of the anolyte. The collective effect of increasing pH and $[K^+]$ increases the production of ethanol (Figure 2f) from $FE_{ethanol}$ 21% in the AEMEA system to 51% in CEMEA system, and we observe the highest $FE_{ethanol}/FE_{ethylene}$ ratio at 0.5 CEL:AEL ratio (Figure 2f).

Ethylene and ethanol shared similar intermediates in the stage immediately following C-C coupling.[61-63] To study the mechanism of this shift in selectivity to ethanol and ethylene as a function of varying local pH and D_{K+}, we used in situ Raman (SERS) across a bulk pH range from 13.5 to 15.0 to explore the adsorption behavior of reactive species on the Cu catalyst. CuO pre-catalyst characterized by SEM, HRTEM, XRD and XPS, showed ~0.5 μm nanoplates (Figure 3a, S8 & S9). The oxidation state of the Cu in the pre-catalyst is mainly 2+, and this is electrochemically reduced to metallic copper during CORR (Figure 3b & S10). We tested the CORR on CuO catalyst at different pH with focus on the phonon modes at ca. 280, 360 and 535 cm^{-1} on SERS (Table S2), corresponding to the restricted rotation of *CO (CO_r), Cu−CO stretching (CO_s) and Cu-OH vibration ($^*OH$).[64-66] The ratio of these peaks we associate with the relative coverage of CO and OH on the copper surface, denoted θ_{CO} and $\theta_{OH,}$ respectively.[35, 67-68] In electrolyte pH range from 13.5 to 14.5, both θ_{OH} and θ_{CO} increases with higher pH, consistent with surface-enhanced infrared measurements (Figure 3d), while the growth of θ_{OH} in low pH ranges may arise due to the lower energy barrier for Cu-OH formation from near neutral to moderate basic pH (Figure 3e).[37-38] The blue shifts of both CO_s and $^*OH$ bands indicate stronger binding of surface CO and OH, along with increased bulk pH values before reaching 14.5 (Figure 3f).

At pH > 14.5, the growth of *CO leveled off, i.e. CO adsorption on the Cu surface saturated (Figure 3d). We saw a significant decrease in θ_{OH} past bulk pH = 14.5 (Figure 3e), a finding we correlate to the decreasing $FE_{ethanol}/FE_{ethylene}$ after that pH (Figure 2g). Previous reports agree on the trend of increased C_{2+} production upon higher CO coverage on the copper surface [38, 67, 69-72], particularly promoting oxygenate formation.[6, 32, 73-76] The influence of alkali cation concentration and pH on selectivity is itself a topic of ongoing discussion in literature.[1, 22, 28, 31-33, 38, 62, 71, 76-78] Recent studies suggest surface OH species is a key promoter of ethanol production.

View Article Online
DOI: 10.1039/D5EE04672F

From our studies, we offer that θ_{CO} and θ_{OH} together may drive ethanol formation: the sharp decrease in θ_{OH} and plateau of θ_{CO} suggest that the surface hydroxide determine product selectivity at high electrolyte concentrations and high pH regions, a model that is able to capture the nonmonotonic dependent of ethanol selectivity on OH/CO coverage ratio[35] and moderate surface OH species for optimal C_{2+} production as documented in previous reports.[79] The absence of Raman shifts on CO_s and $^*OH$ above bulk pH = 14.5 suggests that the integrated effects of [K^+] and [OH^-] do not significantly affect the adsorption of *CO and *OH (Figure 3f). Hence, the lower population of surface-bound hydroxide species can be explained by 1) a high density of K^+ on the electrified interface minimizing the interfacial availability of H_2O/OH;[28-29, 80] and 2) a balance between the higher consumption rate of surface hydroxide and its availability.[36, 68] This is also supported by a recent study that showed a negative correlation between ethanol/ethylene selectivity and concentrated cation (> 6M)),[28] as well as a nonmonotonic distribution of ethanol/ethylene selectivity along similar concentration ranges of pH values.[33, 76]

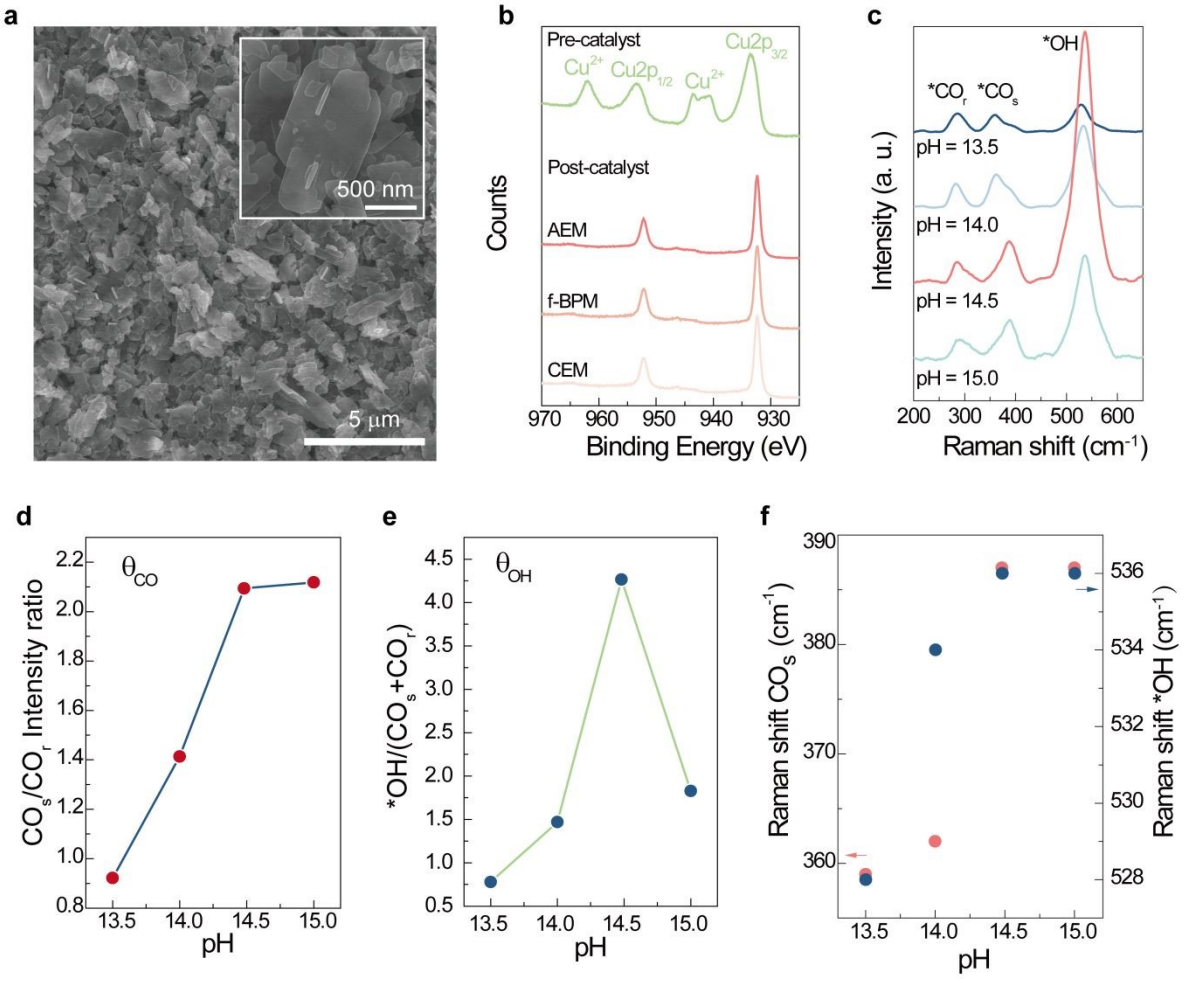


Figure 3. Cathode characterization and in situ Raman analysis. a. Scanning electron microscope image of the CuO pre-catalyst. **b.** High-resolution XPS spectra of the catalyst-coated gas diffusion electrode (GDE) pre- and post-electrolysis. **c.** Operando Raman spectra of CuNPs acquired during CORR at bulk pH 13.5-15.0 at potential

View Article Online
DOI: 10.1039/D5EE04672F

equal to -0.4 V vs. RHE; Relative CO (**d**) and OH (**e**) coverage as a function of the electrolyte pH. **f.** pH-dependent Raman shifts of CO and OH adsorption peaks.

Mechanistic studies

To study further the correlation between the surface hydroxide and ethanol selectivity, we used $H_2{}^{18}O$ solution for the anolyte 1M KOH (^{18}O atom % > 96% in total) and carried out in situ electrochemical mass spectrometry (EC-MS) to track the source of the oxygen in the ethanol (Figure 4c). We tracked the isotope-labelled (48 amu) and unlabelled (46 amu) ethanol and record the percentage of labeled: we obtain 70%, 50% and 42% in f-BPMEA (AEL:CEL = 0.5 ratio), CEMEA and AEMEA, respectively (Figure 4a). This trend aligns well with the nonmonotonic distribution of $FE_{ethanol}/FE_{ethylene}$ upon $pH_{CL/M}$ on this system (Figure 4a), consistent with the nonmonotonic distribution of θ_{OH} along the same pH range (Figure 3b). The $FE_{ethanol}/FE_{ethylene}$ selectivity is positively related to the ratio of $Et^{18}OH$ (Figure 4b), from which we conclude that a favoured route to ethanol formation involves oxygen from the electrolyte instead of from carbon monoxide.

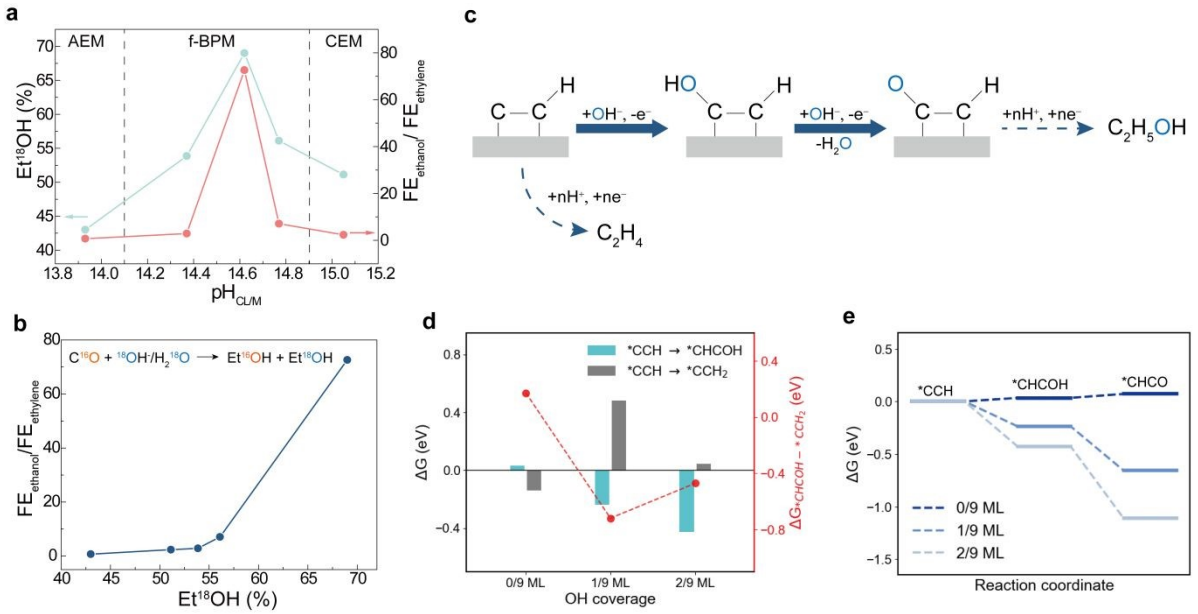


Figure 4. EC-MS isotope analysis and proposed mechanism. **a.** Ethanol/ethylene selectivity and ^{18}O-ethanol percentage as functions of membrane type (corresponding $pH_{CL/M}$) in the MEA setup. **b.** Ethanol/ethylene selectivity versus ^{18}O-ethanol percentage in the MEA setup. **c.** Key reaction pathways for CO reduction (CORR) to ethanol on Cu(111). **d.** The reaction energies for the *CCH→*CHC^{18}OH and *CCH→*CCH_2 steps (left axis) and their corresponding energy differences (right axis, $\Delta G_{*CHCOH-*CCH2}$) at -0.4 V vs SHE at various OH coverage. $\Delta G_{*CHCOH–*CCH2}$ represents the relative preference for *CHCOH or CCH_2 formation, where a higher $\Delta G_{*CHCOH–*CCH2}$ value indicates more favorable *CHCOH formation and consequently higher ethanol selectivity. **e.** Calculated Gibbs free-energy diagram for *CCH to *CHCO on Cu(111) at -0.4 V vs SHE at various OH coverages.

View Article Online
DOI: 10.1039/D5EE04672F

While the $Et^{18}OH$ can arise due to the acetaldehyde pathway[81-83] or due to *CCH hydrolysis[62], the nonmonotonic distribution of ^{18}O-ethanol percentage with monotonically increasing pH, and the insensitivity of labelled/unlabelled propanol in different membrane configurations (Figure S11), indicate little role for the acetaldehyde route (Figure S18). We focus therefore on *CCH[62] as the intermediate determining the selectivity between ethanol and ethylene, and found that increasing OH coverage lowers the energy of *CHCHOH formation (ethanol pathway) from 0.49 eV at 0/9 OH coverage to -0.09 eV at 2/9 OH coverage (Figure S12). However, under ultra-high alkali conditions, the water concentration at the interface is expected to be much lower than OH^-, suggesting that the oxygen in ethanol should originate mainly from surface-bound or solvated hydroxide. Hence, the observation of ^{18}O-labeled ethanol indicates that one of the intermediates may be oxidized by the hydroxyl-containing species, also supported by a recent study that found that CO is oxidized by *OH under high coverage of CO and OH on copper.[36] To account for the potential oxidation step, we propose the conversion of *CCH to *CHCOH to form ^{18}O-labelled ethanol or to $*CCH_2$, leading to ethylene (Figure 4c). As shown in Figure 4d, higher OH coverage enhances *CHCOH formation, while the formation energy of $*CCH_2$ initially increases and then decreases. 1/9 ML OH coverage exhibits the lowest energy difference between *CHCOH and $*CCH_2$ (Figure S13-S15). Although *CHCOH selectivity does not increase monotonically with OH coverage, the negative $\Delta G_{*CHCOH-*CCH_2}$ agrees with the preference for *CHCOH formation in the presence of surface *OH (Figure 4d). DFT calculations herein also show that *OH coverage reduces the energy required for *CHCO formation (Figure 4d), which is energetically favorable compared to the hydrolysis step of *CCH to *CHCHOH (Figure S9),[62] promoting ethanol formation over ethylene.

Conclusions

This work studies how membrane configurations tune product selectivity in CORR. The correlation between system and reaction selectivity was linked through membrane kinetics and interfacial speciation at molecular level. Among the configurations tested, a f-BPM delivers >93% selectivity for liquid production and 70-to-1 ethanol selectivity over ethylene by maintaining optimal local pH and potassium ion availability at the catalyst surface. The higher CEL-to-AEL ratio will lead to elevated pH values between CL and the membrane in the reported systems. These conditions favor pathways that suppress gaseous products ethylene and hydrogen. Operando techniques, coupled with computational modeling, indicate that *CO and *OH surface coverage drive selectivity toward ethanol over ethylene. Future work could beneficially probe the interfacial pH in such MEA systems.[84]

Overall, the present findings highlight the importance of membrane design in catholyte-free MEA systems for efficient carbon utilization and the potential of f-BPM for liquid production.

This work focuses on the use of f-BPMEA for CORR, although CO_2RR could, in principle, be

View Article Online
DOI: 10.1039/D5EE04672F

performed in the same device. In this case, CO_2RR in f-BPMEA will need to tackle the challenge of salt precipitation at the cathode, this the result of the high local concentration of cations near the catholyte interface; and CO_2 bubble formation at the CEL/AEL junction, and its potential to produce delamination of the f-BPM, would need also to be addressed.[45, 85]

Methods

Chemicals

Copper(II) sulfate pentahydrate ($CuSO_4 \cdot 5H_2O$), Methanol (ACS reagent grade, >99.8%), Dimethyl Sulfoxide (ACS reagent grade >99.9%), Lithium hydroxide (ACS reagent grade 98%), Sodium hydroxide (ACS reagent grade 98%), Potassium hydroxide (semiconductor grade 99.99%), water-^{18}O (97 atom % ^{18}O) were purchased from Sigma Aldrich. Nafion solution (D520, 5wt% and D2020, 20wt%), Carbon paper (Freudenberg H23C6), iridium oxide, Nafion 212 and Nafion 211 were purchased from the Fuel cell store. Sustainion (X37-50 grade 60) was purchased from Dioxide Materials. All chemicals were used as received or pretreated under standard procedures. The aqueous solutions were prepared using de-ionized water with a resistivity of 18.2MΩcm.

Synthesis of Cu catalysts

The CuO nanoplate material was prepared by hydrothermal method modified from previous protocol.[32] In a typical procedure, 1.0 g $CuSO_4 \cdot 5H_2O$ were dissolved in 50 ml deionized water to form a homogeneous blue solution. The solution was placed in an ice-water bath with vigorous magnetic stirring. 10 ml 1.2M NaOH solution was dropped into the above solution slowly and stirred half an hour continually. Then, keep the mixture refrigerated at 3 ^oC for 24 hours before transferring it to the Teflon-lined autoclave. The hydrothermal program was set to 130 ^oC for 18 hours, followed by cooling to room temperature. The Cu performed material was collected via centrifugation, washed several times with deionized water and dried in vacuum at 60 ^oC overnight.

Materials Characterization

The morphologies of samples were characterized by SEM (Hitachi SU8030). TEM characterization was conducted using a JEOL ARM-200F microscope equipped with a JEOL delta aberration corrector and a 200 kV cold field emission gun with powder samples. X-ray powder diffractometer (XRD) measurements were performed in a MiniFlex600 with Cu- Kα radiation. XPS measurements were conducted using the ThermoFisher Scientific NEXSA G2, equipped with monochromated, micro-focused, low-power Al K-Alpha X-ray source. Catalyst coated electrodes were used directly for measurements.

Preparation of gas diffusion electrodes (GDEs)

Cu NPs were dispersed in a mixture of methanol and Nafion solution and left under

View Article Online
DOI: 10.1039/D5EE04672F

ultrasonication for at least 30min. The catalyst ink composition was 25 mg_{CuNPs} ml_{MeOH}^{-1} and a 100 μl_{Nafion} 5wt% ml_{MeOH}^{-1}. The ink was spray coated on carbon paper GDL with a Cu nanoparticle loading of 3 mg cm^{-2} to prepare the Cu electrode. The electrodes were left under room conditions overnight before the electrochemical measurements were conducted.

Preparation of Bipolar Membranes

CEM with different thickness was fabricated following a reported method.[53] 0.5 ml Nafion® D2020 dispersion was dispensing and spin-coating onto pre-cleaned, and dust-free Si wafer chips (~6 cm^2) at 500-5000 rpm. The samples were then dried on a hot stage at 80 °C for 60 s. Alternatively, the thicker membrane was fabricated by casting polymer solutions on Si wafer by a homemade using a doctor blade with glass rod and double sided tape. The samples were then transferred into a tube furnace and annealed in N_2 for 4 h at 110 °C. The thickness and uniformity of each membranes were measured with a profilometer on the center and different edges to obtain <5% thickness difference. The edges of the CEM on the Si were removed with a razor, allowing the film to fully detach from the substrate in the next step. The substrate with the thin film were slowly immersed into water and the Sustainion AEM was carefully pressed onto the layer. Then the thin-CEL BPM was removed from the water bath and transferred to the MEA. This method was used to produce BPMs with 2-15 μm CEM, while the Nafion 211 and Nafion 212 were used for BPMs with 25- and 50-mm thickness CEM. The BPMs were used immediately after assembly.

Electrochemical measurements

A homemade titanium MEA electrolyzer was used, consisting of a gas chamber and an anodic chamber. The MEA was composed of a cathode electrode, an Ni foam (or NiFe-P) anode electrode and an ion-exchange membrane (Sustainion for AEMEA, Nafion for CEMEA, BPM for BPMEA). The electrochemical testing cell was assembled by placing the membrane between the cathode and anode electrode, in which the catalyst layers of both electrodes were facing the membrane. The electrodes were protected and surrounded by 0.01-inch-thick gaskets for electrical insulation. The reaction area was regulated by the pore area in the gasket, which was set at 1 cm^2. This assembly was then inserted into the homemade fuel cell device, which had serpentine flow channels that were equally compressed with torque applied to the bolts. A constant rate of 20 ml min^{-1} of KOH anolyte flowed through the anodic channel, and a digital mass flow meter supplied the humidified CO feed gas to the cathodic channel at a constant rate of 25 sccm. After the electrolyzer assembly, a galvanostatic method was used to apply electrolysis. For the stability test, the MEA was operated at a constant full-cell current density of 100 mA cm^{-2} with NiFe-P anode.

Electrochemical reduction product measurement

The gas products were collected from the gas outlet of the MEA and injected into a gas

View Article Online
DOI: 10.1039/D5EE04672F

chromatograph (PerkinElmer Clarus 600) coupled with a flame ionization detector (FID) for the detection of CH_4 and C_2H_4 and a thermal conductivity detector (TCD) for the detection of H_2, and CO signals. The Faradaic efficiency of gas products was calculated using the following equation:

$$FE_i = y_i V z_i F \frac{P_0}{RT} j_{total}$$

where FE_i is the Faradaic efficiency of product "i", y_i is the volumetric fraction of the gas product, V is the outlet gas flow rate in sccm, z_i is the number of electrons associated with producing one molecule of product from CO, F is the Faraday constant, P_0 is the atmospheric pressure, R is the ideal gas constant, T is the temperature and j_{total} is the total current density.

The liquid products were analyzed using ^{1}H NMR spectroscopy (500 MHz Bruker Avance III HD) with water suppression. Dimethyl sulfoxide was used as the reference standard and deuterium oxide as the lock solvent. The Faradaic efficiency of liquid products was calculated using the equation:

$$FE_i = n_i z_i F \frac{1}{Q_{total}}$$

where n_i is the number of moles of liquid product "i" and Q_{total} is the total charge passed through the cell prior to liquid sampling.

Operando spectroscopic analysis

Operando Raman spectroscopy was operated with a water immersion objective (×63) using a Renishaw inVia Raman microscope. The spectra were collected using a 785 nm laser at 0.1% intensity. Each spectrum was recorded using the Renishaw WiRE (v.4.4) software by integrating three times, with each integration lasting 10 s. An open-structured flow cell was utilized for the measurements. CO was continuously supplied to the gas chamber during the measurement in a modified flow cell set up with standard Cu NPs on GDE as working electrode. An Ag/AgCl electrode (filled with saturated aqueous KCl solution) and a platinum wire were used as the reference and counter electrode, respectively (Figure S16). While -0.4 V vs. RHE was applied to the Raman cell, electrolyte with different pH was exchanged by flow to monitor the surface species on the same laser spot. Actual pH was measured by pH meter. The potentials from the Raman measurements were converted to values versus RHE.

EC-MS

In situ EC-MS analysis was conducted using a Hiden HPR-20 QIC capillary system coupled with a customized electrochemical cell. The gaseous outlet of regular MEA setup with different membrane configurations was connected to the EC-MS. 1 M KOH solution was prepared using water-^{18}O (97 atom % ^{18}O). The gaseous products were continuously monitored

View Article Online
DOI: 10.1039/D5EE04672F

simultaneously under electrochemical potential protocols that varied in real time. All gaseous products entered the capillary inlet and were then ionized. Ionization was performed at an electron energy of 70V. The resulting ions were detected using a secondary electron multiplier (SEM) detector operating at 800V.

Computation

Spin-polarized density functional theory (DFT) calculations[86] were performed using the Vienna Ab Initio Simulation Package (VASP)[87-88]. The exchange-correlation interactions in the Kohn-Sham equations were modeled using the generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional[89-90]. Core-valence interactions were described through the projector-augmented wave (PAW) method[91], while long-range dispersion effects were incorporated using Grimme's DFT-D3(BJ) scheme[92-93]. An energy cutoff of 450 eV was employed, and the Brillouin zone was sampled with a 4 × 4 × 1 Monkhorst-Pack k-point grid[94]. Structural optimization was carried out until the residual forces on atoms were below 0.03 eV/Å and energy changes were less than 10^{-4} eV. The free energies were calculated based on the computational hydrogen electrode (CHE) model[95].

A 3×3 Cu(111) supercell comprising four atomic layers was constructed as the surface model, with the bottom two layers fixed in their bulk lattice positions. To account for solvation effects, three layers of water containing 18 water molecules were placed on the Cu(111) surface. The local pH on the Cu(111) surfaces with OH coverages of 0/9, 1/9, and 2/9 ML were thus approximated as 7, 14.47, and 14.74, respectively. Additionally, a vacuum layer of at least 15 Å was introduced perpendicular to the surface to minimize interactions between periodic images.

Data availability

All data are available from the authors on reasonable request.

Acknowledgements

This work was supported by Bill & Melinda Gates Foundation under the investment # INV-061828 and INV-079283. This work made use of facilities at IMSERC at Northwestern University.

References

[1] M. Jouny, W. Luc, F. Jiao, *Nat. Catal.* **2018**, *1*, 748-755.
[2] P. Zhu, C. Xia, C. Y. Liu, K. Jiang, et al., *Proc Natl Acad Sci U S A* **2021**, *118*.
[3] W. Niu, J. Feng, J. Chen, L. Deng, et al., *Nat Commun* **2024**, *15*, 7070.
[4] A. Ozden, Y. Wang, F. Li, M. Luo, et al., *Joule* **2021**, *5*, 706-719.
[5] H. Shin, K. U. Hansen, F. Jiao, *Nat. Sustain.* **2021**, *4*, 911-919.

View Article Online
DOI: 10.1039/D5EE04672F

[6] F. Li, Y. C. Li, Z. Wang, J. Li, et al., *Nat. Catal.* **2019**, *3*, 75-82.
[7] M. Luo, Z. Wang, Y. C. Li, J. Li, et al., *Nat Commun* **2019**, *10*, 5814.
[8] J. Ding, H. Bin Yang, X.-L. Ma, S. Liu, et al., *Nat. Energy* **2023**, *8*, 1386-1394.
[9] L. Zhang, J. Feng, L. Wu, X. Ma, et al., *J. Am. Chem. Soc.* **2023**, *145*, 21945-21954.
[10] Z. Liu, L. Song, X. Lv, M. Liu, et al., *J. Am. Chem. Soc.* **2024**.
[11] C. Long, K. Wan, Y. Chen, L. Li, et al., *J. Am. Chem. Soc.* **2024**, *146*, 4632-4641.
[12] X. Wang, P. Ou, A. Ozden, S.-F. Hung, et al., *Nat. Energy* **2022**, *7*, 170-176.
[13] K. V. Petrov, C. I. Koopman, S. Subramanian, M. T. M. Koper, et al., *Nat. Energy* **2024**, *9*, 932-938.
[14] D. A. Salvatore, C. M. Gabardo, A. Reyes, C. P. O'Brien, et al., *Nat. Energy* **2021**, *6*, 339-348.
[15] T. Burdyny, W. A. Smith, *Energy Environ. Sci.* **2019**, *12*, 1442-1453.
[16] M. A. Blommaert, R. Sharifian, N. U. Shah, N. T. Nesbitt, et al., *J Mater Chem A Mater* **2021**, *9*, 11179-11186.
[17] K. Yang, M. Li, S. Subramanian, M. A. Blommaert, et al., *ACS Energy Lett* **2021**, *6*, 4291-4298.
[18] D. A. Henckel, P. Saha, S. Rajana, C. Baez-Cotto, et al., *ACS Energy Lett* **2024**, *9*, 3433-3439.
[19] F. P. Garcia de Arquer, C. T. Dinh, A. Ozden, J. Wicks, et al., *Science* **2020**, *367*, 661-666.
[20] E. Perez-Gallent, G. Marcandalli, M. C. Figueiredo, F. Calle-Vallejo, et al., *J. Am. Chem. Soc.* **2017**, *139*, 16412-16419.
[21] A. S. Malkani, J. Li, N. J. Oliveira, M. He, et al., *Sci Adv* **2020**, *6*.
[22] S. J. Shin, H. Choi, S. Ringe, D. H. Won, et al., *Nat Commun* **2022**, *13*, 5482.
[23] Y. Xu, Z. Xia, W. Gao, H. Xiao, et al., *Nat. Catal.* **2024**, *7*, 1120-1129.
[24] S. Yu, H. Yamauchi, S. Wang, A. Aggarwal, et al., *Nat. Catal.* **2024**, *7*, 1000-1009.
[25] Z.-M. Zhang, T. Wang, Y.-C. Cai, X.-Y. Li, et al., *Nat. Catal.* **2024**, *7*, 807-817.
[26] Q. Zhu, C. L. Rooney, H. Shema, C. Zeng, et al., *Nat. Catal.* **2024**, *7*, 987-999.
[27] J. M. Yoo, J. Ingenmey, M. Salanne, M. R. Lukatskaya, *J. Am. Chem. Soc.* **2024**.
[28] H. Zhang, J. Gao, D. Raciti, A. S. Hall, *Nat. Catal.* **2023**, *6*, 807-817.
[29] N. T. Nesbitt, W. A. Smith, *The Journal of Physical Chemistry C* **2021**, *125*, 13085-13095.
[30] H. Xiao, T. Cheng, W. A. Goddard, 3rd, R. Sundararaman, *J. Am. Chem. Soc.* **2016**, *138*, 483-486.
[31] C. T. Dinh, T. Burdyny, M. G. Kibria, A. Seifitokaldani, et al., *Science* **2018**, *360*, 783-787.
[32] X. Wang, Y. Chen, F. Li, R. K. Miao, et al., *Nat Commun* **2024**, *15*, 616.
[33] W. Luc, X. Fu, J. Shi, J.-J. Lv, et al., *Nat. Catal.* **2019**, *2*, 423-430.
[34] J. Zhang, G. Zeng, S. Zhu, H. Tao, et al., *Proc Natl Acad Sci U S A* **2023**, *120*, e2218987120.
[35] A. Herzog, M. Lopez Luna, H. S. Jeon, C. Rettenmaier, et al., *Nat Commun* **2024**, *15*, 3986.
[36] H. Zhang, Y. Cui, C. Ren, Q. Li, et al., *ACS Catal.* **2024**, *14*, 10737-10745.
[37] X. Chang, Y. Zhao, B. Xu, *ACS Catal.* **2020**, *10*, 13737-13747.
[38] J. Li, X. Chang, H. Zhang, A. S. Malkani, et al., *Nat Commun* **2021**, *12*, 3264.
[39] M. Etzi Coller Pascuzzi, A. J. W. Man, A. Goryachev, J. P. Hofmann, et al., *Catalysis Science & Technology* **2020**, *10*, 5593-5601.
[40] Y. Han, J. Wang, Y. Liu, T. Li, et al., *Carbon Neutralization* **2024**, *3*, 172-198.
[41] D. Prando, A. Brenna, M. V. Diamanti, S. Beretta, et al., *J Appl Biomater Funct Mater* **2017**, *15*, e291-e302.
[42] R. M. Katona, J. Carpenter, E. J. Schindelholz, R. F. Schaller, et al., *J. Electrochem. Soc.* **2021**, *168*.

[43] Q. Ye, X. Zhao, R. Jin, F. Dong, et al., *J. Mater. Chem. A* **2023**, *11*, 21498-21515.
[44] C. P. O'Brien, R. K. Miao, A. Shayesteh Zeraati, G. Lee, et al., *Chem. Rev.* **2024**, *124*, 3648-3693.
[45] X. She, L. Zhai, Y. Wang, P. Xiong, et al., *Nat. Energy* **2024**, *9*, 81-91.
[46] Y. C. Li, Z. Yan, J. Hitt, R. Wycisk, et al., *Advanced Sustainable Systems* **2018**, *2*.
[47] K. Xie, R. K. Miao, A. Ozden, S. Liu, et al., *Nat Commun* **2022**, *13*, 3609.
[48] C. P. O'Brien, R. K. Miao, S. Liu, Y. Xu, et al., *ACS Energy Lett.* **2021**, *6*, 2952-2959.
[49] T. Alkayyali, A. S. Zeraati, H. Mar, F. Arabyarmohammadi, et al., *ACS Energy Lett.* **2023**, *8*, 4674-4683.
[50] J. C. Bui, I. Digdaya, C. Xiang, A. T. Bell, et al., *ACS Applied Materials & Interfaces* **2020**, *12*, 52509-52526.
[51] B. Mayerhöfer, D. McLaughlin, T. Böhm, M. Hegelheimer, et al., *ACS Applied Energy Materials* **2020**, *3*, 9635-9644.
[52] J. A. Wrubel, Y. Chen, Z. Ma, T. G. Deutsch, *J. Electrochem. Soc.* **2020**, *167*.
[53] S. Z. Oener, L. P. Twight, G. A. Lindquist, S. W. Boettcher, *ACS Energy Lett.* **2020**, *6*, 1-8.
[54] W. L. Toh, H. Q. Dinh, A. T. Chu, E. R. Sauvé, et al., *Nat. Energy* **2023**, *8*, 1405-1416.
[55] T. Alkayyali, M. Zargartalebi, A. Ozden, F. Arabyarmohammadi, et al., *Joule* **2024**, *8*, 1478-1500.
[56] C. McCallum, C. M. Gabardo, C. P. O'Brien, J. P. Edwards, et al., *Cell Reports Physical Science* **2021**, *2*.
[57] T. Yamanaka, T. Takeguchi, H. Takahashi, W. Ueda, *J. Electrochem. Soc.* **2009**, *156*.
[58] Z. Jing, Y. Zhou, T. Yamaguchi, K. Yoshida, et al., *J Phys Chem Lett* **2023**, *14*, 6270-6277.
[59] J. C. Bui, E. W. Lees, D. H. Marin, T. N. Stovall, et al., *Nature Chemical Engineering* **2024**, *1*, 45-60.
[60] G. A. El-Nagar, F. Haun, S. Gupta, S. Stojkovikj, et al., *Nat Commun* **2023**, *14*, 2062.
[61] F. Calle-Vallejo, M. T. Koper, *Angew. Chem. Int. Ed.* **2013**, *52*, 7282-7285.
[62] Y. Lum, T. Cheng, W. A. Goddard, 3rd, J. W. Ager, *J. Am. Chem. Soc.* **2018**, *140*, 9337-9340.
[63] R. Kortlever, J. Shen, K. J. Schouten, F. Calle-Vallejo, et al., *J Phys Chem Lett* **2015**, *6*, 4073-4082.
[64] C. M. Gunathunge, X. Li, J. Li, R. P. Hicks, et al., *The Journal of Physical Chemistry C* **2017**, *121*, 12337-12344.
[65] G. Niaura, *Electrochim. Acta* **2000**, *45*, 3507-3519.
[66] B. D. Smith, D. E. Irish, P. Kedzierzawski, J. Augustynski, *J. Electrochem. Soc.* **1997**, *144*, 4288-4296.
[67] C. Zhan, F. Dattila, C. Rettenmaier, A. Bergmann, et al., *ACS Catal* **2021**, *11*, 7694-7701.
[68] C. Zhan, F. Dattila, C. Rettenmaier, A. Herzog, et al., *Nat. Energy* **2024**.
[69] R. B. Sandberg, J. H. Montoya, K. Chan, J. K. Nørskov, *Surface Science* **2016**, *654*, 56-62.
[70] X. Chang, H. Xiong, Q. Lu, B. Xu, *JACS Au* **2023**, *3*, 2948-2963.
[71] A. Wuttig, C. Liu, Q. Peng, M. Yaguchi, et al., *ACS Cent Sci* **2016**, *2*, 522-528.
[72] M. Schreier, Y. Yoon, M. N. Jackson, Y. Surendranath, *Angew. Chem. Int. Ed.* **2018**, *57*, 10221-10225.
[73] Y. Ji, Z. Chen, R. Wei, C. Yang, et al., *Nat. Catal.* **2022**, *5*, 251-258.
[74] J. Jin, J. Wicks, Q. Min, J. Li, et al., *Nature* **2023**, *617*, 724-729.
[75] P. Wei, D. Gao, T. Liu, H. Li, et al., *Nat Nanotechnol* **2023**, *18*, 299-306.
[76] J. Li, Z. Wang, C. McCallum, Y. Xu, et al., *Nat. Catal.* **2019**, *2*, 1124-1131.

View Article Online
DOI: 10.1039/D5EE04672F

[77] J. Li, D. Wu, A. S. Malkani, X. Chang, et al., *Angew. Chem. Int. Ed.* **2020**, *59*, 4464-4469.
[78] Y. Cao, Z. Chen, P. Li, A. Ozden, et al., *Nat Commun* **2023**, *14*, 2387.
[79] M. Sun, A. Staykov, M. Yamauchi, *ACS Catal.* **2022**, *12*, 14856-14863.
[80] X. Yang, H. Ding, S. Li, S. Zheng, et al., *J. Am. Chem. Soc.* **2024**, *146*, 5532-5542.
[81] Y. Hori, R. Takahashi, Y. Yoshinami, A. Murata, *The Journal of Physical Chemistry B* **1997**, *101*, 7075-7081.
[82] E. Bertheussen, A. Verdaguer-Casadevall, D. Ravasio, J. H. Montoya, et al., *Angew. Chem. Int. Ed.* **2016**, *55*, 1450-1454.
[83] E. L. Clark, A. T. Bell, *J. Am. Chem. Soc.* **2018**, *140*, 7012-7020.
[84] Y. Y. Wang, H. Shi, Y. Gong, B. Zhang, et al., *Langmuir* **2023**, *39*, 16807-16811.
[85] J. Disch, S. Ingenhoven, S. Vierrath, *Advanced Energy Materials* **2023**, *13*.
[86] W. Kohn, L. J. Sham, *Physical Review* **1965**, *140*, A1133-A1138.
[87] G. Kresse, D. Joubert, *Physical Review B* **1999**, *59*, 1758-1775.
[88] G. Kresse, J. Furthmüller, *Physical Review B* **1996**, *54*, 11169-11186.
[89] J. P. Perdew, K. Burke, M. Ernzerhof, *Physical Review Letters* **1996**, *77*, 3865-3868.
[90] J. P. Perdew, J. A. Chevary, S. H. Vosko, K. A. Jackson, et al., *Physical Review B* **1992**, *46*, 6671-6687.
[91] P. E. Blöchl, *Physical Review B* **1994**, *50*, 17953-17979.
[92] S. Grimme, J. Antony, S. Ehrlich, H. Krieg, *The Journal of Chemical Physics* **2010**, *132*, 154104.
[93] S. Grimme, S. Ehrlich, L. Goerigk, *Journal of Computational Chemistry* **2011**, *32*, 1456-1465.
[94] H. J. Monkhorst, J. D. Pack, *Physical review B* **1976**, *13*, 5188.
[95] J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, et al., *J. Phys. Chem. B* **2004**, *108*, 17886-17892.

View Article Online
DOI: 10.1039/D5EE04672F

Judd A. and Marjorie Weinberg College of Arts and Sciences
Northwestern University

Department of Chemistry
2145 Sheridan Road
Evanston, Illinois 60208-3113

Ted Sargent
Professor of Chemistry

ted.sargent@northwestern.edu

Data Availability Statement

All data are available from the authors on reasonable request.