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# Bi<sub>2</sub>O<sub>3</sub>/nylon multilayered nanocomposite membrane for the photocatalytic inactivation of waterborne pathogens and degradation of mixed organic pollutants†

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A powder semiconductor  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> was synthesized *via* solid state thermal annealing and further composited with a multilayered nylon fibrous membrane *via* electrospaying. The successful integration of nano-sized  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> into the nylon membrane was confirmed by XRD, FESEM images, UV-vis, FT-IR and Raman analyses. The nanocomposite membrane displayed a visible light-responsive catalytic ability with an energy bandgap of 2.78 eV estimated from the data of DRS. The activity of the composite membrane was examined in a continuous mode reactor for the degradation of separate and mixed solutions of anionic (indigo carmine) and cationic (rhodamine B) organic pollutants. Moreover, the composite membrane exhibited antibacterial properties towards *E. coli*, a waterborne pathogen, as revealed by the obtained growth inhibition during the Kirby–Bauer and liquid culture tests. The inactivation of *E. coli* was confirmed by live/dead cell staining using fluorescence imaging. Finally, a mixed solution of organic dyes at different concentration, as well as the recycle of the membrane, showed the stability and potential of the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite membrane for the removal of organic pollutants and inactivation of the waterborne pathogen.

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## Environmental significance

Worldwide, there is an increasing demand for clean water and sanitation systems and different solutions are under evaluation, including advanced oxidation processes such as photocatalysis. This work describes the scalable synthesis process of an electrospun composite membrane made of nylon and embedded  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> nanoparticles that can be activated by visible light instead of UV light typically used with other nanomaterials (*e.g.* TiO<sub>2</sub>). As a proof of concept, the efficacy of the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> electrospun composite membrane in the visible light inactivation of pollutants and pathogens was demonstrated in a continuous-flow photocatalytic membrane reactor, highlighting the great potential of this advanced photocatalytic process for clean water and sanitation.

## 1. Introduction

Nowadays, there is an increasing demand for clean water and sanitation. One of the emerging technologies to get sustainable clean water can be based on advanced oxidation processes (AOPs). Among them, heterogeneous photocatalysis is an AOP in which semiconductor materials are illuminated with solar/UV light to facilitate redox reactions and enable the treatment of organic, inorganic species, and microbes.<sup>1,2</sup> However, its suitability in real applications is still challenged

by various essential aspects. One of these is the cost-effective recovery of dispersed particles, their filtration and recycling.<sup>3–5</sup> Indeed, for high photoactivity, nano-scale particles are preferred because of their prominent surface area, better dispersion in bulk reactors, and optimum activation from the irradiation source.<sup>6,7</sup> However, these particles are hard to recover and reuse; moreover, they increase the operational cost of water treatment.<sup>3,4</sup> Alternatively, various studies have suggested diverse materials as fixed supports for the active particles, with a focus on carbon, silica, and mineral-based porous supports that could allow a good contact between the immobilized semiconductors and the targeted contaminant for continuous treatment. Examples of such supports are silica-based,<sup>8–10</sup> activated carbon,<sup>11,12</sup> synthetic clay LAPONITE®,<sup>13</sup> glass, steel mesh,<sup>14–16</sup> and recently some polymers such as polyimide.<sup>17</sup> However, with such immobilized fixed supports, the

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photocatalytic response is reportedly lower than using bulk materials, because of low dispersion, reduced transparency, limited mass transfer, and weak interaction of the immobilized semiconductor with the targeted contaminants.<sup>3</sup> Thus, these limitations hinder the advancement of non-slurry application alternatives, as the immobilization supports should satisfy requirements of chemical inertness, transparency to the irradiation source, and adequate channels and permeation to capture the targeted contaminants in bulk reactors.<sup>18,19</sup>

Merging membrane and photocatalytic technologies has been considered recently. In this case, the semiconductor materials could be incorporated/embedded within the porous membrane to maximize the degradation efficiency. Such an approach turned into a great achievement in treating wastewater. The porous network of tiny channels in the membrane could be used to embed the semiconductor materials, and further to capture and adsorb the contaminants, so that their removal could be maximized. Recent results have shown that adding semiconductor particles to the membrane has improved water flux,<sup>20</sup> refined permeability and selectivity,<sup>21</sup> boosted antifouling properties<sup>22</sup> and increased adsorption sites, which ultimately enhanced the adsorption of the membrane.<sup>23</sup> Different kinds of materials were used to prepare photocatalytic membranes such as ceramics, zeolites, but mostly polymers.<sup>24</sup> However, irradiation and generation of oxidizing species in the reaction environment cause abrasion and leaching of the semiconductor from the membrane itself.<sup>24</sup> Therefore, for its chemical, thermal, and mechanical stability, nylon polymer could be a promising option for semiconductor embedding and composite usage as a photocatalytic membrane. Indeed, electrospun nylon-6 nanofibers have a large surface area and active sites that could improve the adsorption of pollutants for the subsequent degradation.<sup>25</sup> Moreover, the electrospun nanofibers have demonstrated improved properties in dye removal,<sup>26</sup> advanced filtration,<sup>27</sup> and antibacterial applications.<sup>28</sup>

Besides the degradation and removal of organic and inorganic pollutants, various studies have explored the potential of semiconductor-polymeric composite membranes for the inactivation and removal of pathogens through the same principle *i.e.* the attack of the cell membrane through reduction-oxidation reactions and consequently reduction in their growth.<sup>28,29</sup> Some of the reported semiconductor composite membranes include embedded oxides of silver, titanium, cerium, zirconium, and iron, and have shown antimicrobial response against common pathogens *i.e.* *Escherichia coli*, *Staphylococcus aureus*, *Enterococcus* sp., *Pseudomonas aeruginosa*, and *Klebsiella pneumoniae*.<sup>30-33</sup> Amongst various semiconductors, bismuth-based materials, such as BiVO<sub>4</sub>,<sup>34</sup> Bi<sub>2</sub>O<sub>3</sub>,<sup>35,36</sup> BiOX (Cl, I, Br),<sup>37</sup> and bismuth-based composites,<sup>38,39</sup> have been proven efficient in water treatment due to their ability to produce reactive oxygen species (ROS). Among them, Bi<sub>2</sub>O<sub>3</sub> has emerged as the most effective visible light-responsive material with promising optical, electrical, thermal, and photocatalytic properties.<sup>40</sup>

Moreover, it also has a tunable bandgap between 2.1 and 2.8 eV compared to 3.2 eV of TiO<sub>2</sub>. The solar light viable bandgap is a result of the Bi<sub>2</sub>O<sub>3</sub> hybrid valence band arising from contributions of Bi 6s and O 2p orbitals; in contrast, the TiO<sub>2</sub> valence band only has O 2p orbital contributions.<sup>41</sup>

To the best of our knowledge, Bi<sub>2</sub>O<sub>3</sub>-polymeric nanocomposite membranes have rarely been investigated against the removal of both organic pollutants and waterborne pathogens. Therefore, this study was conducted to exploit and investigate the potential of Bi<sub>2</sub>O<sub>3</sub> integration into a nylon-6 membrane. Once synthesized, the membranes were characterized by XRD, FESEM, UV-vis, FT-IR and Raman analyses, to check the proper integration of the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> nanomaterials into the nylon membrane. Afterward, the degradation efficiency of the composite membranes against single and mixed solutions of anionic (indigo carmine) and cationic (rhodamine-B) organic dye pollutants and against a pathogenic strain of *E. coli* was evaluated by using a continuous-flow photocatalytic membrane reactor (CPMR), paving the way for a new approach in achieving water purification.

## 2. Experimental section

### 2.1 Materials and chemicals

Nylon-6 pellets (density of 1.084 g mL<sup>-1</sup>), formic acid, ethanol, polyethylene glycol (PEG), Triton-X, bismuth(III) nitrate pentahydrate (Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O), indigo carmine (IC), and rhodamine B (RhB) were purchased from Sigma Aldrich, Italy. Sodium chloride (NaCl) was purchased from Daejung Co. Ltd., Korea. The bacterial culture media (agars and broths) were purchased from Oxoid, England. For live/dead cell staining, a LIVE/DEAD® BacLight™ Bacterial Viability kit was purchased from ThermoFisher Scientific, USA; this kit included two staining dyes, SYTO-9 and propidium iodide (PI), at 20 mM concentration each. All the materials and chemicals were used as received.

### 2.2 Synthesis of $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub>

The  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> powder was synthesized from Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O by solid state thermal annealing route. The synthesis details are published elsewhere.<sup>42</sup> In brief, the measured quantity of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O was placed in a ceramic crucible and directly heated inside a muffle furnace at an initial temperature of 150 °C for 30 min. Afterward, the temperature was raised to 250 °C and kept constant for 2 h, and finally, the temperature was increased to 550 °C for 45 minutes to obtain the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite. This composite (in bulk form) was already reported as a visible light-responsive semiconductor for the degradation of organic compounds.<sup>42</sup>

### 2.3 Preparation of the Bi<sub>2</sub>O<sub>3</sub>/nylon composite membrane

The obtained  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> powder (10 wt%) was dissolved in ethanol. Measured drops of PEG and Triton-X were gradually added into the ethanol mixture under continuous mixing *i.e.* to ensure a well-segregated particle suspension and

homogenization. The obtained solution was stirred for 18 hours and then sonicated for 30 minutes to get a homogeneous yellow colored solution. The nylon-6 solution was made by dissolving 22 wt% nylon pellets in formic acid, stirring it continuously at 100 rpm for 6 hours to produce a neat and transparent nylon-6 solution. Then, the nylon-6 solution was electrospun to produce a multilayered nylon fibrous membrane. The multilayers referred to the electrospun and stacked fibrous layers of nylon that overlaid one after another. The applied voltage was altered between 18 and 24 kV to obtain thick and ultra-thin spiderweb type multilayered nano-fibers.<sup>43</sup> The injection flow rate was set to 0.95 mL h<sup>-1</sup> at a 12 cm tip to collector distance. Electrospinning was followed by electro-spray of the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> solution on the surface of the obtained nylon electrospun multilayers *i.e.* for successful integration of  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub>. Fig. 1 depicts the scheme of the electrospinning process for the preparation of bare and  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite membranes and the digital image of the obtained membrane. The obtained membranes were referred to as bare-nylon and Bi<sub>2</sub>O<sub>3</sub>/nylon composite. Visually, the surface of the obtained Bi<sub>2</sub>O<sub>3</sub>/nylon membrane showed a homogeneous pale-yellow non-dusty appearance as an indication of the embedded Bi<sub>2</sub>O<sub>3</sub> particles.

#### 2.4 Characterization of the bare and composite membranes

To observe and analyze the microstructure and morphology of the obtained bare and composite membranes, an in-lens detector of a Zeiss Supra 40 field emission scanning electron microscope (FESEM) was used to acquire secondary electron contrast images with 5 keV electrons (Zeiss SMT, Oberkochen, Germany). Energy-dispersive X-ray spectroscopy (EDX) analysis was performed during FESEM imaging. To this aim, all the peaks were used for the elemental analysis (3 iterations), which was carried out with 15 keV electrons. To analyze the size and phase composition of the obtained

membranes, a Panalytical X'Pert MRD Pro Cu K $\alpha$  X-ray source in Bragg/Brentano configuration was used. The sample was scanned in a  $2\theta$  range of 20–60°. A Varian Cary 500 spectrophotometer equipped with an integration sphere was used to record the diffuse reflectance spectroscopy (DRS) and transmittance spectra of the bare and composite membranes. Tauc plots of the Kubelka–Munk function were used to calculate the energy bandgap ( $E_g$ ) value. FT-IR analysis was performed in attenuated total reflection (ATR) mode to identify the chemical changes in the membrane after the incorporation of  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub>. The spectra were collected with a 4 cm<sup>-1</sup> resolution by accumulating 64 scans each spectrum on a Nicolet 5700 FTIR spectrometer (Thermo Fisher) equipped with a diamond crystal and a DTGS detector (at room temperature) in the 400–4000 cm<sup>-1</sup> range. Raman spectra were acquired with a 785 nm laser source using an InVia Qontor Raman microscope (Renishaw plc, Wotton-under-Edge, UK) equipped with a 20 $\times$  objective, in backscattering configuration. The output laser power was attenuated to 25 mW and the acquisition time was 50 s.

#### 2.5 Photocatalytic evaluation

**2.5.1 Continuous-flow photocatalytic membrane reactor (CPMR).** Fig. 1 shows the schematic diagram of the in-house fabricated CPMR for the evaluation of the efficacy of the membranes on selected dyes and bacteria. The CPMR includes a cylindrical tube made of borosilicate glass with four membrane holders (each 6  $\times$  6 cm) for placing the prepared membrane in a straight position at a distance of 10 cm from each other. A white LED light source for the activation of integrated  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub>, with a measured irradiance of around 70 W m<sup>-2</sup> (at the cylinder's center of all positioned membranes) and an emission spectrum between 400 and 700 nm, was attached outside of the cylinder. Beneath the cylindrical tube, the storage tank was connected *via* a

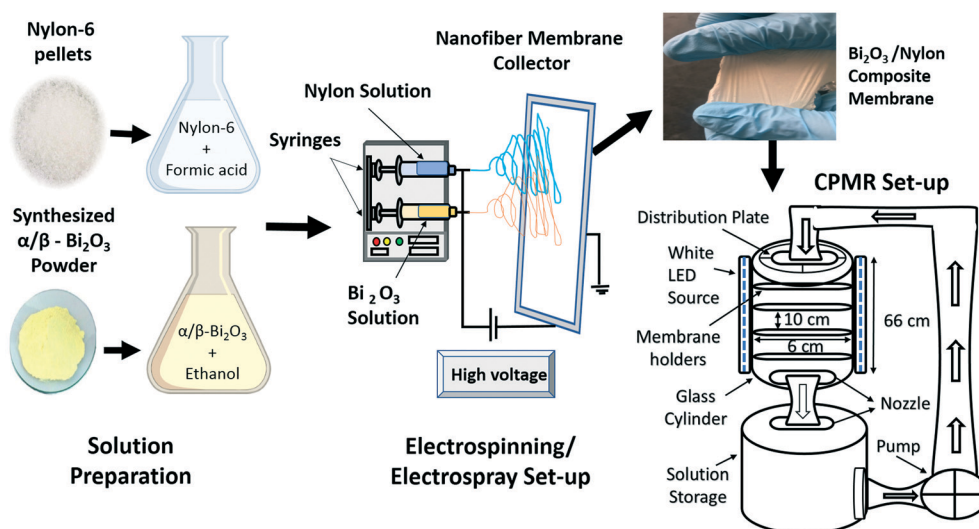


Fig. 1 Schematic layout of membrane preparation and photocatalytic tests.

peristaltic pump *i.e.* for the circulation of the stock solution (2 L) over the membrane (*i.e.* inside the cylindrical tube) at a constant flow of 25 mL min<sup>-1</sup>. For dark conditions, the LED light remained switched-off; besides, aluminum foil was wrapped on the cylindrical tube.

**2.5.2 Photocatalytic discoloration tests.** Anionic IC and cationic RhB dyes were selected as model pollutants because of their common use in photocatalytic evaluation.<sup>42</sup> For the preliminary experiments, both dyes were tested separately at a concentration of 5 ppm. Initially, to achieve adsorption/desorption equilibrium, the LED light remained switched-off for the first 30 minutes. Afterward, the solution was irradiated to analyze the photocatalytic removal and assess the degradation kinetics. 3 mL of the samples was collected in a predefined time sequence of photocatalytic treatment, centrifuged for 5 minutes for the analysis, and afterward returned to the stock solution to avoid any interference with the system kinetics. The centrifuged samples were analyzed using a double beam UV-vis spectrophotometer (Perkin Elmer-LAMBDA 365) for recording the time-dependent intensity reduction and changes in the absorbance spectra of the model pollutants. For the mineralization degree of the treated dye solutions, total organic carbon (TOC) analysis was carried using a TOC-L SHIMADZU TOC analyzer. For TOC analysis, the control and the treated samples of both dyes were collected and analyzed using a protocol reported in the literature.<sup>44</sup> The stability of the membrane was evaluated by testing the same membrane up to three cycles using the IC solution at a fixed concentration of 5 ppm. Before reuse, the recovered membranes were thoroughly rinsed with distilled water.

For the composite membrane performance against the mixed regime of dyes, *i.e.* of different chemical structures and ionic behaviors, separately prepared solutions of IC and RhB (at a molar concentration of  $1 \times 10^{-6}$  M) were mixed at even proportions. The prepared mixed stock of 2 L was evaluated in the CPMR, under similar operating conditions, as mentioned earlier.

**2.5.3 Photocatalytic bacterial inhibition/inactivation.** Initially, the inhibition of *E. coli* ATCC (8739) was performed on agar plates, using the bare and composite membranes cut into  $10 \times 10$  mm<sup>2</sup> pieces. In brief, the inoculum (100  $\mu$ L) was taken from the overnight incubated *E. coli* culture at a concentration of  $1 \times 10^6$  CFU mL<sup>-1</sup> and was spread on LB agar plates. Then, the bare nylon/composite membrane was placed at the center of the plate and incubated for 24 h at  $37 \pm 1$  °C; the test was performed in the dark or in the presence of a white LED lamp (100 W m<sup>-2</sup>). After 24 hours, the inhibition zone was observed in the culture plates.<sup>45</sup> The experiments were conducted in duplicate, and the zone of inhibition was calculated as suggested in previously reported studies.<sup>46,47</sup> Further, for the evaluation of the liquid bacterial suspension, *E. coli* was inoculated in a 50 mL tube of LB broth (Oxoid, England) after overnight incubation at 37 °C and shaking at 120 rpm. Afterward, the culture broth was centrifuged at 5000 rpm for 10 minutes, and the settled pellets were washed several times with a sterilized normal

saline solution (0.85% of NaCl). Then, the bacterial biomass was diluted in 2 L of saline solution to reach a bacterial concentration of around  $1 \times 10^6$  CFU mL<sup>-1</sup>, by measuring an optical density (OD) of 0.01 at 600 nm. The prepared bacterial stock was tested in the CPMR for photocatalytic inactivation, at 25 mL min<sup>-1</sup>. Initially, the dark conditions were maintained for 30 minutes, then the cylindrical tube was irradiated. A 100  $\mu$ L sample was collected in a predefined time sequence of photocatalytic treatment. The collected samples were serially diluted and plated on LB agar plates, incubated for 24 hours at 37 °C, and the numbers of bacterial colonies were counted.

To observe if the photocatalytic action was able to damage the *E. coli* cells, live/dead bacterial cell fluorescence staining was performed. For this, a 1 mL sample was collected at different time intervals, centrifuged at 10000 rpm for 15 minutes, and then the bacterial biomass was washed with sterilized saline solution. Finally, the obtained bacterial biomass was resuspended in 1 mL of sterilized saline solution. Afterward, by following the protocol provided in the received live/dead fluorescence staining kit, a 1:1 mixed dye stock solution of SYTO-9 (3.34 mM, excitation 483 nm, emission 503 nm) and propidium iodide (PI; 20 mM, excitation 535 nm, emission 617 nm) was prepared. For fluorescence microscopy imaging, 5  $\mu$ L of the mixed dye stock solution was added into the bacterial suspension and incubated in the dark for 30 minutes. After incubation, 5  $\mu$ L of the stained bacterial suspension was collected and pipetted over a glass slide and covered for fluorescence analysis. A fluorescence microscope (Zeiss Axio Scope.A1, Carl Zeiss, Germany) equipped with FITC and Texas RED filters was used to acquire fluorescence images. The acquired images were analyzed *via* ImageJ 1.50d to assess the percentage of live (green-stained) and dead (red-stained) bacterial cells.

## 3. Results and discussion

### 3.1 Characterization

The composite membrane containing the bismuth oxide-based nanomaterial was synthesized and characterized, as described in the experimental part. In the following, the physical-chemical characterization using XRD, UV-vis, FT-IR, Raman spectroscopy, FESEM imaging, EDX analysis is reported.

**3.1.1 XRD analysis.** The XRD patterns of bare nylon, Bi<sub>2</sub>O<sub>3</sub>/nylon, and synthesized  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> powder are shown in Fig. 2A. The bare nylon, being semi-crystalline, does not show any peak, while in the case of the Bi<sub>2</sub>O<sub>3</sub>/nylon composite some peaks were observed at 27.06°, 31.7°, and 53.6°, indicating the formation of the monoclinic  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> phase.<sup>42</sup> These peaks match with standard XRD data (JCPDS 16-0654).<sup>43</sup> Additional peaks were observed at  $2\theta = 27.96$ , 32.72°, and 46.22°, corresponding to the  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> phase (JCPDS card no. 01-078-1793).<sup>42</sup> As shown by the comparison in Fig. 2A, the pattern of the Bi<sub>2</sub>O<sub>3</sub>/nylon composite retained the main peaks observed for the  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> powder after integration into nylon by electrospinning. However, due to the dilution

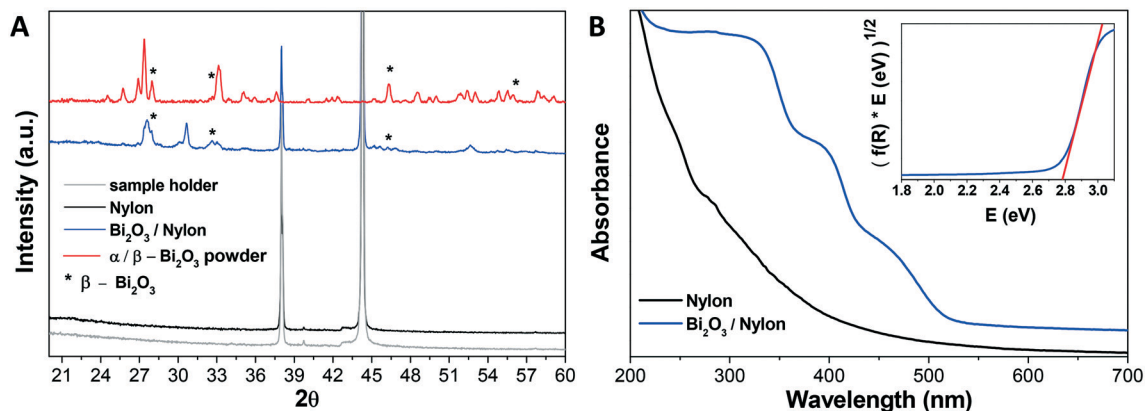


Fig. 2 A) XRD patterns of the sample holder, bare nylon, Bi<sub>2</sub>O<sub>3</sub>/nylon composite, and synthesized α/β-Bi<sub>2</sub>O<sub>3</sub> powder, and B) UV-vis DRS spectra of bare nylon and Bi<sub>2</sub>O<sub>3</sub>/nylon composite; inset: Tauc plot for the estimation of the bandgap.

in the nylon membrane, the weaker peaks in the XRD spectra of α/β-Bi<sub>2</sub>O<sub>3</sub> could not be detected.<sup>48</sup>

**3.1.2 Optical properties.** Fig. 2B shows the DRS spectra of the bare and composite nylon membranes. The bare nylon displayed a slight absorption in the region 250 nm to 300 nm, typical of nylon polymer.<sup>49</sup> Meanwhile, the Bi<sub>2</sub>O<sub>3</sub>/nylon composite membrane exhibited one absorbance stretch in the plateau between 375 nm and 450 nm, which is the characteristic optical transition range of α-Bi<sub>2</sub>O<sub>3</sub>. Moreover, the extended absorbance stretch in the range of 450–530 nm represented the characteristic transition range of β-Bi<sub>2</sub>O<sub>3</sub>.<sup>42</sup> The observed absorption edge in the visible spectrum could facilitate the transition of the electrons from the valence to the conduction band and consequently allow the composite membrane to harvest solar irradiation for photocatalytic activity.<sup>34</sup> The bandgap of the composite membrane estimated from the Tauc plot was around 2.78 eV, *i.e.* shown in the inset

of Fig. 2B. As per the XRD analysis, the α-phase is dominant in the α/β-Bi<sub>2</sub>O<sub>3</sub> composite. Therefore, the two absorption edges in the visible region have a combined representation in the Tauc plot, and the predominant characteristics of α-Bi<sub>2</sub>O<sub>3</sub> were slightly narrowed due to the presence of β-Bi<sub>2</sub>O<sub>3</sub>.

**3.1.3 FESEM analysis.** Fig. 3 shows the FESEM images of bare nylon (Fig. 3A and B) and the Bi<sub>2</sub>O<sub>3</sub>/nylon composite (Fig. 3C and D) at different magnifications. In the case of bare nylon, the images exhibit both thick and thin electrospun nanofiber strands in the range of 20–60 nm diameter that form a net-like structure with overlaid multilayers.<sup>43,50</sup> The obtained nanofibers displayed a uniform and bead-free formation. In the case of the Bi<sub>2</sub>O<sub>3</sub>/nylon composite, the FESEM image in Fig. 3C shows the integration of nano-sized round Bi<sub>2</sub>O<sub>3</sub> particles into the nylon fibers. Moreover, Fig. 3D suggests that this integration was achieved across each electrospun layer and in the whole thickness of the membrane.

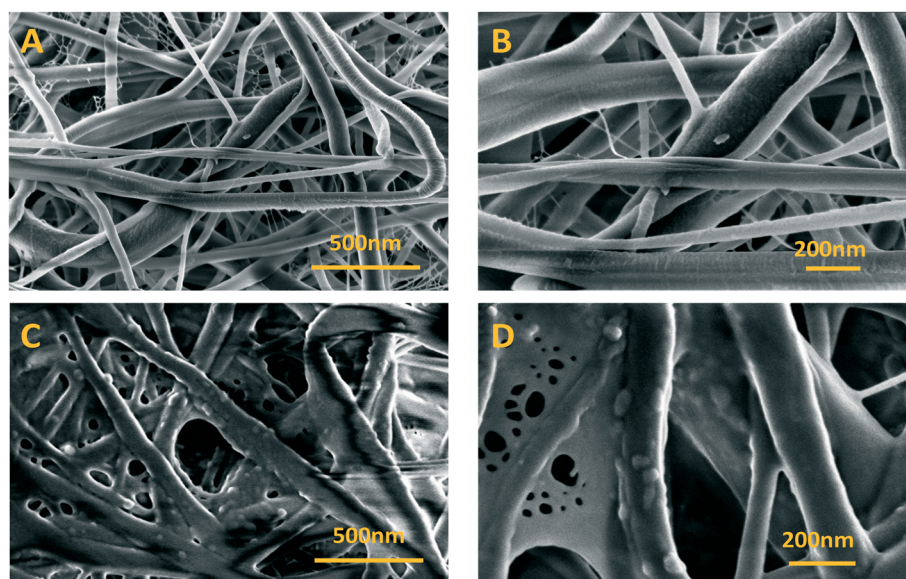


Fig. 3 FESEM images of bare nylon and Bi<sub>2</sub>O<sub>3</sub>/nylon membranes at different magnifications. (A) Bare nylon (25k×), (B) (50k×), (C) Bi<sub>2</sub>O<sub>3</sub>/nylon (25k×), and (D) (50k×).

A further confirmation of the integration of the bismuth nanomaterial into the multilayered electrospun membrane was revealed by EDX analysis, reported in Fig. S1.† This evaluation clearly revealed the presence of Bi atoms in the  $\text{Bi}_2\text{O}_3$ /nylon membrane.

**3.1.4 FT-IR analysis.** Fig. 4 shows the FT-IR spectra of the bare and composite membrane samples. Nylon-6 is a polyamide; therefore, its molecular structure is characterized by the presence of amide groups as displayed by the amide I stretching at  $1638\text{ cm}^{-1}$  ( $\text{C}=\text{O}$  stretching) and amide II stretching at  $1540\text{ cm}^{-1}$  (as a combination of  $-\text{CN}$  stretching and  $-\text{NH}$  bending).<sup>51</sup> At the same time, the intense band at  $3297\text{ cm}^{-1}$  indicates the N-H stretching vibration.<sup>52</sup> The peaks at about  $1367\text{ cm}^{-1}$ ,  $1263\text{ cm}^{-1}$ , and  $1200\text{ cm}^{-1}$  are due to amide III stretching and  $\text{CH}_2$  wagging, while the peaks at  $2860\text{ cm}^{-1}$  and  $2933\text{ cm}^{-1}$  are due to  $\text{CH}_2$  symmetric and asymmetric stretching, respectively.<sup>52</sup> All other peaks match with FT-IR assignments of nylon-6 nanofibers reported previously and summarized in Table S1.†<sup>51,52</sup> In the  $\text{Bi}_2\text{O}_3$ /nylon composite, concerning  $\text{Bi}_2\text{O}_3$ , its leading bands are below  $500\text{ cm}^{-1}$ , but the ATR diamond crystal used to acquire the spectra cuts this region. The only clear signal of bismuth oxide is the peak at  $846\text{ cm}^{-1}$  due to the symmetric Bi-O stretching.<sup>53</sup> Nevertheless, some of the absorption bands of nylon-6 are downshifted. In particular, the amide I and amide II bands are shifted to  $1635\text{ cm}^{-1}$  and  $1538\text{ cm}^{-1}$ , respectively, whereas the N-H stretching is shifted to  $3294\text{ cm}^{-1}$ . This is probably the result of the interaction of the nylon polymer with  $\text{Bi}_2\text{O}_3$ .

**3.1.5 Raman analysis.** Raman analysis of the bare nylon and composite membranes was performed to further assess the incorporation of the  $\alpha/\beta\text{-Bi}_2\text{O}_3$  nanoparticles. As shown in Fig. 5 the nylon fibers are characterized by a weak Raman signal that matches the typical vibrational pattern of polyamides.

In detail, different bands assigned to CC stretching coupled to  $\text{CH}_2$  bending modes are observed around  $1100\text{ cm}^{-1}$  (CC stretching +  $\text{CH}_2$  twisting around  $1080\text{ cm}^{-1}$  and CC stretching

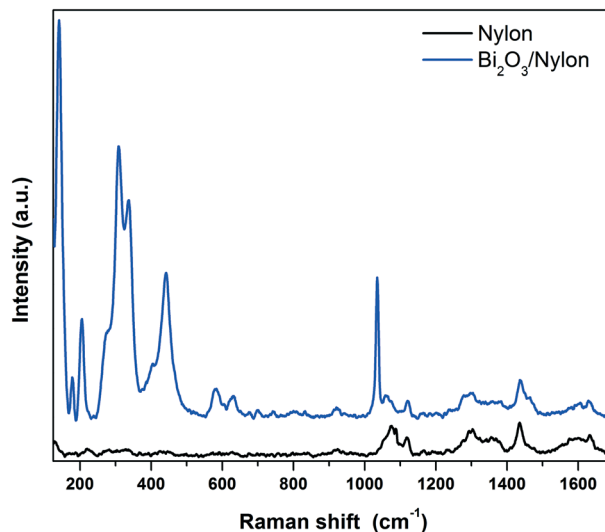


Fig. 5 Raman spectra of the bare (bottom black curve) and  $\alpha/\beta\text{-Bi}_2\text{O}_3$  decorated (top blue curve) nylon membranes.

+  $\text{CH}_2$  wagging at  $1120\text{ cm}^{-1}$ ), while two  $\text{CH}_2$  twisting bands and one  $\text{CH}_2$  bending-related band are detected between  $1260$  and  $1300\text{ cm}^{-1}$  and at  $1440\text{ cm}^{-1}$ , respectively.<sup>54</sup> Finally, the amide I band appears at about  $1635\text{ cm}^{-1}$ .<sup>55</sup> The addition of the  $\text{Bi}_2\text{O}_3$  nanopowder results in significant changes in the low Raman shift range, while the vibrational pattern is almost unaffected over  $600\text{ cm}^{-1}$ , except for a narrow band at  $1040\text{ cm}^{-1}$  due to residual nitrates from the synthesis. The emergence of intense Raman modes in the  $120\text{--}600\text{ cm}^{-1}$  range, attributable to two different  $\text{Bi}_2\text{O}_3$  crystalline phases, clearly evidences the incorporation of the bismuth oxide in the membrane. The bands at  $278$ ,  $313$  and  $440\text{ cm}^{-1}$  were previously assigned to  $\alpha\text{-Bi}_2\text{O}_3$ .<sup>56,57</sup> Meanwhile, the vibrational modes at  $142$  and  $335\text{ cm}^{-1}$  appear slightly up-shifted with respect to the main bands of  $\beta\text{-Bi}_2\text{O}_3$  ( $128\text{ cm}^{-1}$  and  $317\text{ cm}^{-1}$ ),<sup>58</sup> but can still be consistent with the presence of the

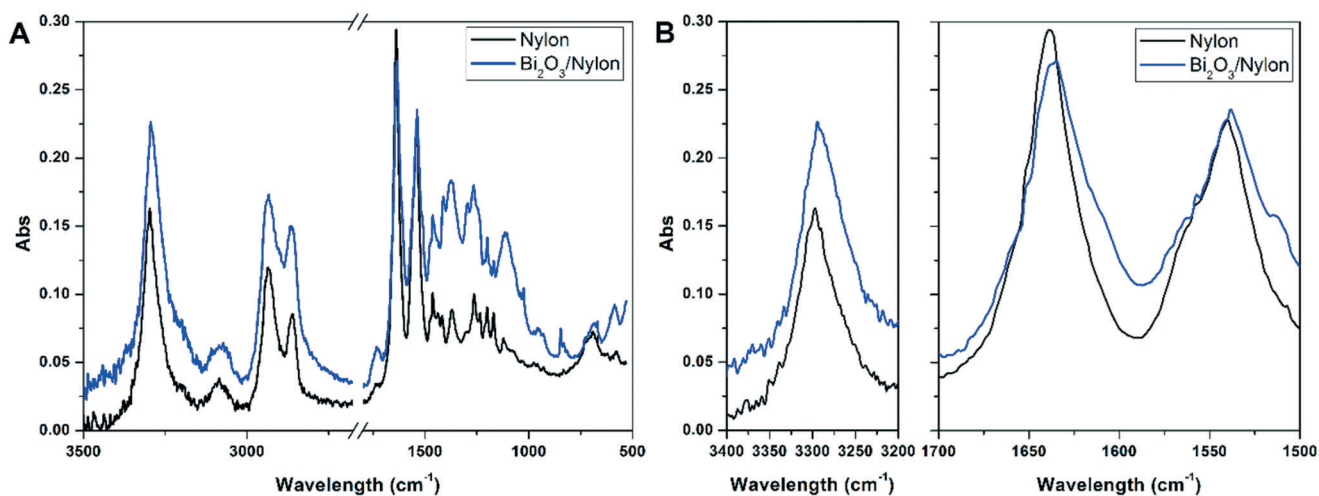


Fig. 4 FT-IR patterns of nylon and  $\text{Bi}_2\text{O}_3$ /nylon membranes: A) whole FT-IR spectrum; B) magnification of the N-H stretching (left) and amide I and amide II (right) regions.

tetragonal phase. Depending on the Raman shift region where they occur, the observed Raman bands can be attributed to Bi-dominated lattice movements ( $<155\text{ cm}^{-1}$ ), modes with significant Bi and O contributions ( $155\text{--}255\text{ cm}^{-1}$ ) and oxygen related displacements ( $>255\text{ cm}^{-1}$ ), as derived previously for  $\alpha\text{-Bi}_2\text{O}_3$ .<sup>57</sup> Raman analysis thus confirms both the formation of the composite and the preservation of the crystalline phases of the nanopowder after electrospinning.

### 3.2 Photocatalytic dye discoloration/degradation

Once the composite membrane has been physico-chemically characterized, its photocatalytic efficiency was evaluated, employing photocatalytic tests against anionic IC and cationic RhB organic dyes. The selection of these dyes was made to compare the obtained results with previously reported ones, in which both dyes were evaluated individually and in a mixed solution using the bulk  $\alpha/\beta\text{-Bi}_2\text{O}_3$  powder.<sup>42</sup>

**3.2.1 Indigo carmine (IC) dye.** Fig. 6A shows the relative concentration ( $C/C_0$ ) profile of IC dye at different times of photocatalytic exposure, using bare nylon and  $\text{Bi}_2\text{O}_3/\text{nylon}$  composite membranes. The absorption spectra of IC vs.

different treatment times using  $\text{Bi}_2\text{O}_3/\text{nylon}$  are given in Fig. S2 of the ESI.† Initially, under dark conditions, the absorbance of IC dye at 610 nm was reduced by 21.4% due to its charge interactions with the nylon multilayers.<sup>59</sup> Any further decrease due to adsorption was not observed for a longer period in the dark. Instead, after irradiation exposure, the absorbance peak of IC started to decrease further, due to the photocatalytic response of  $\alpha/\beta\text{-Bi}_2\text{O}_3$ , until it completely disappeared after prolonged exposure of 180 minutes. The removal of IC is due to the oxidation and breaking of the indigoid group ( $\text{NHC}=\text{CNH}$ ), as indicated by the observed isosbestic point at 250 nm and by the increase of absorbance of intermediate products (shown in Fig. S2†) *i.e.* isatin sulfonic acid and 2-amine-5-sulfo-benzoic acid, in the UV absorbance region between 250 nm and 275 nm.<sup>5,42</sup> The persistent appearance of these peaks was probably an indication of incomplete mineralization.<sup>44</sup>

Further, in Fig. 6A the photolysis of IC dye solution, *i.e.* under irradiation exposure without the presence of any membrane, showed no/less significant change in the IC concentration, while in the case of the bare nylon membrane, around 19% of IC dye was initially adsorbed in the dark and,

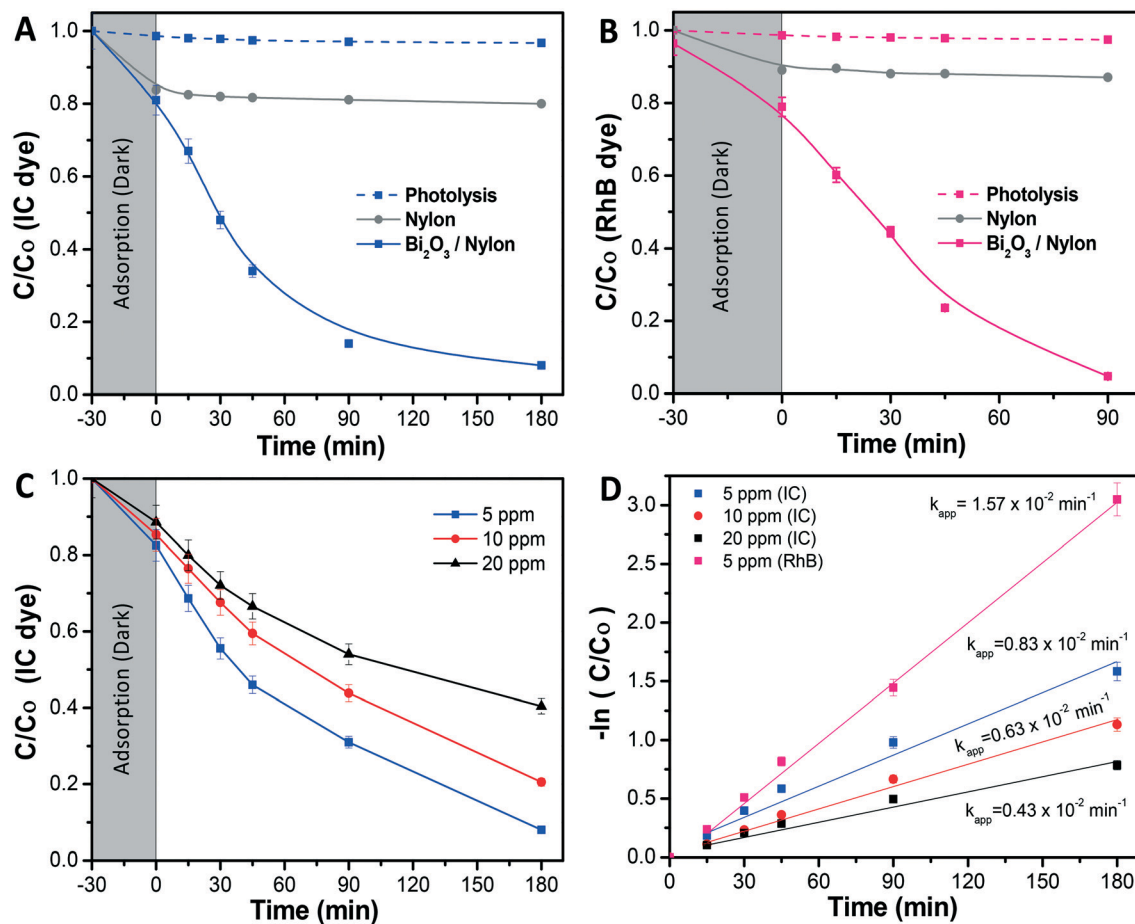


Fig. 6 Photocatalytic tests using bare nylon and  $\text{Bi}_2\text{O}_3/\text{nylon}$  composite membranes: A) relative concentration ( $C/C_0$ ) of IC degradation vs. time ( $C_0 = 5\text{ ppm}$ ); B)  $C/C_0$  of RhB degradation vs. time ( $C_0 = 5\text{ ppm}$ ); C)  $C/C_0$  of different IC dye concentrations with the  $\text{Bi}_2\text{O}_3/\text{nylon}$  composite membrane; and D) kinetic curve of IC and RhB dye solutions, and estimated apparent kinetic rates ( $k_{app}$ ).

afterward, did not show any further activity under light conditions. This could be ascribed to a limited amount of adsorption sites in the nylon fibrous membrane, which could not accommodate further dye molecules, resulting in a limited removal of the dye from the solution.<sup>26</sup> Meanwhile, the Bi<sub>2</sub>O<sub>3</sub>/nylon composite membrane showed slightly higher adsorption (21.4%), compared to the bare nylon membrane. This could be related to additional adsorption sites provided by the embedded nanoparticles.<sup>23</sup> Afterward, in the following first 45 min of irradiation, the improved removal of IC was observed due to the high concentration of adsorbed IC molecules on the membrane surface, which was rapidly oxidized by the generated ROS after irradiation. Further, after 180 min of irradiation, around 78% of mineralization was achieved, as analyzed by TOC analysis. Afterward, the TOC reduction remained unchanged for a longer run of 360 min, due to the persistent nature of the resulting intermediates.<sup>42</sup> The kinetic curve for the degradation of IC (shown in Fig. 6D; at 5 ppm concentration) is plotted using the  $\ln(C_0/C)$  function vs. time which resulted in a straight line ascribed to the pseudo-first-order reaction,<sup>44</sup> with a calculated apparent kinetic rate ( $K_{app}$ ) of  $0.83 \times 10^{-2} \text{ min}^{-1}$ .

**3.2.2 Rhodamine B (RhB) dye.** RhB is a cationic dye and has both diethylamine and carboxylic groups.<sup>42</sup> Fig. S3 in the ESI† shows the UV-vis absorbance spectra of RhB at different time intervals obtained with the Bi<sub>2</sub>O<sub>3</sub>/nylon composite membrane. RhB shows the maximum absorbance peak at 555 nm,<sup>40</sup> which was followed to relate its removal *via* simultaneous adsorption and photocatalytic oxidation. In Fig. 6B, the  $C/C_0$  profiles of RhB dye at different times of photocatalytic exposure, using bare nylon and Bi<sub>2</sub>O<sub>3</sub>/nylon composite membranes, are compared. Initially, 21% of the RhB intensity was reduced due to its interaction with the multilayers and integrated Bi<sub>2</sub>O<sub>3</sub> particles *i.e.* under dark conditions, and the adsorption–desorption equilibrium was attained with the saturation of the composite membrane by RhB dye molecules as no further adsorption was noted up to 60 min under dark conditions. The interaction of cationic dyes with nylon is assumed to be weaker than that of anionic dyes. However, an improved interaction of RhB dye with the composite membrane was observed. This could be associated with the anionic carboxylic group of RhB and good protonation of  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> particles that could have allowed amphoteric sites of RhB to interact with the composite membrane.<sup>42</sup> After irradiation, the adsorbed and highly interacting RhB dye was removed and oxidized by the reactive species generated by the composite  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub> particles after 90 minutes, and reached a degradation and mineralization degree of up to 68% after 180 min of exposure, as observed through TOC analysis. The mineralization of RhB was unchanged after that due to the formation of persistent intermediates.<sup>44</sup> Compared to IC dye, the removal of RhB was found to be faster with a calculated  $K_{app}$  of  $1.57 \times 10^{-2} \text{ min}^{-1}$  *i.e.* shown in Fig. 6D. The improved kinetics of RhB could be due to its high affinity towards the membrane and the close interaction with the ROS produced by the composite  $\alpha/\beta$ -

Bi<sub>2</sub>O<sub>3</sub> particles. Moreover, Fig. 6B shows no significant change in the RhB concentration in the case of treatment with the bare nylon membrane, under photolysis conditions.

**3.2.3 Varied dye concentrations.** To evaluate the membrane performance at high loading of the IC dye, the photocatalytic tests were conducted at varied IC concentrations using the Bi<sub>2</sub>O<sub>3</sub>/nylon composite. Fig. 6C and D show the  $C/C_0$  and removal kinetics of IC at concentrations of 5, 10 and 20 ppm, and reveal that the removal efficiency of the Bi<sub>2</sub>O<sub>3</sub>/nylon composite membrane decreased to 75.11% and 61.45% at concentrations of 10 and 20 ppm, respectively, during 180 minutes of irradiation exposure. This behavior could be related to the low availability/generation of ROS due to the attachment of the dye molecules on the active sites of  $\alpha/\beta$ -Bi<sub>2</sub>O<sub>3</sub>. Secondly, the resulting increase in solution opacity and consequently the decreased irradiation transmission subsequently lowered the removal efficiency of IC.<sup>60,61</sup> Moreover, the high organic load could reduce the water flux and increase the mass transfer resistance of the membrane and could decrease the membrane efficiency.<sup>62</sup> This trend was observed during TOC analysis, which revealed a decrease in the degradation and mineralization efficiency of the composite membrane, *i.e.* after 180 min of irradiation, the analyzed TOC reduction was 63% and 37% for 10 and 20 ppm concentrations of IC dye, respectively, compared to 78% reduction in the case of 5 ppm solution. The resultant decrease in mineralization efficiency could be due to the saturation of the composite membrane with intermediate species that affected the semiconductor activation to support the generation of reactive species and induce oxidation. Fig. 6D shows calculated  $K_{app}$  values of  $0.63 \times 10^{-2} \text{ min}^{-1}$  and  $0.43 \times 10^{-2} \text{ min}^{-1}$  for 10 and 20 ppm IC dye concentrations, significantly reduced compared to  $0.83 \times 10^{-2} \text{ min}^{-1}$  in the case of 5 ppm concentration.

**3.2.4 Mixed dye degradation.** Since colored wastewater contains mixtures of several dyes, interfering with each other in the mixed effluent, their removal behavior and mechanism of interaction for any potential treatment (including the case of membrane-assisted photocatalysis) could be different. Therefore, the evaluation of a mixed solution of both IC and RhB dyes was performed using the composite membrane. The recorded absorbance spectrum of the mixed solution vs. irradiation time given in Fig. 7A shows two main peaks corresponding to the contribution of both IC and RhB dyes at 610 and 555 nm, respectively. In Fig. 7B, the  $C/C_0$  profiles of the mixed solution at different times of photocatalytic exposure are reported *i.e.* using bare nylon and Bi<sub>2</sub>O<sub>3</sub>/nylon composite membranes.

Under dark conditions, both IC and RhB dyes preferentially adsorbed on the membrane at 23% and 22%, respectively. After irradiation, the absorbance of the IC dye was significantly reduced in the first 90 min (Fig. 7B), with the generated intermediate products absorbing in the UV region *i.e.* from 200 nm to 250 nm. Due to an intense peak (Fig. 7A), RhB dye depletion followed IC removal, but at slightly lower kinetics. Actually, the calculated  $K_{app}$  was  $1.05 \times 10^{-2} \text{ min}^{-1}$  and  $1.21 \times 10^{-2} \text{ min}^{-1}$  in the case of RhB and IC,



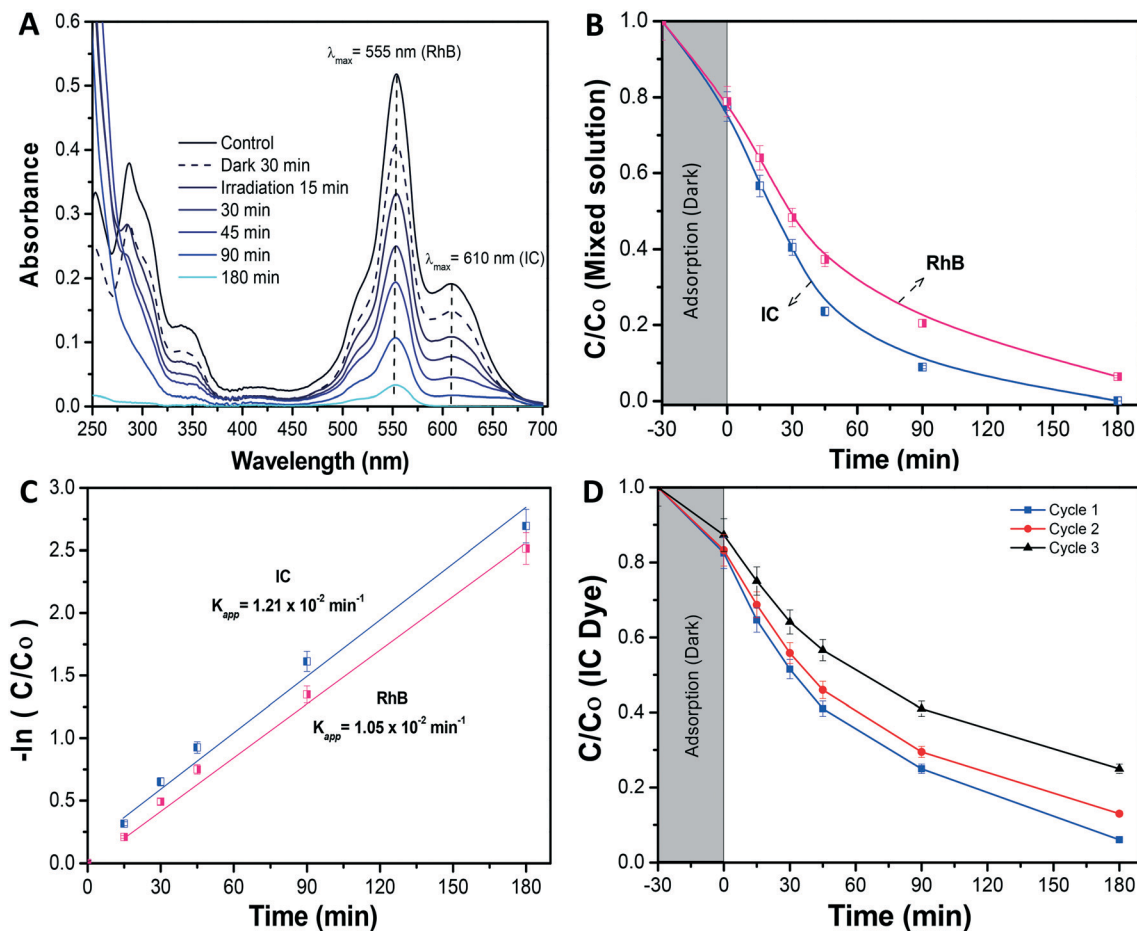


Fig. 7 A) Absorbance spectra vs. irradiation time of the mixed solution, B) relative concentration ( $C/C_0$ ) of combined dyes degraded using the  $\text{Bi}_2\text{O}_3/\text{nylon}$  composite, C) kinetic curves of removal of IC and RhB dyes in the mixed solution, and estimated  $K_{\text{app}}$ , and D) cyclic stability of the  $\text{Bi}_2\text{O}_3/\text{nylon}$  composite membrane up to three cycles;  $C/C_0$  profiles at 5 ppm IC dye concentration.

respectively. The degradation kinetics of RhB was also found to be lower compared to when it was tested alone ( $K_{\text{app}} 1.57 \times 10^{-2} \text{ min}^{-1}$ ) (Fig. 7C). This decrease in RhB removal kinetics probably was due to the limited ROS availability for the attack and degradation of both dyes. The investigation revealed that both dyes underwent simultaneous degradation with the formation of IC dye intermediate products probably because of the good interaction of both dyes enabled by the membrane, in contrast to our previously reported observation with the dispersed  $\alpha/\beta\text{-Bi}_2\text{O}_3$  nanoparticles, where IC was preferentially degraded, and after its complete degradation, the activity was directed against RhB.<sup>42</sup>

Here the equal and concurrent interaction of both dyes with the generated ROS is supported by the membrane. The overall obtained results suggested no/less interfering behavior among the tested dyes, as the removal kinetics of both dyes was almost similar *i.e.* from  $1.21 \times 10^{-2} \text{ min}^{-1}$  and  $1.05 \times 10^{-2} \text{ min}^{-1}$ , for IC and RhB, respectively.

**3.2.5 Cyclic stability.** For the cyclic stability of the  $\text{Bi}_2\text{O}_3/\text{nylon}$  nanocomposite membrane, the photocatalytic tests were carried out up to three cycles *i.e.* after the composite membrane was recovered, rinsed and configured again in the

CPMR. Fig. 7D shows the relative concentration ( $C/C_0$ ) decrease of the IC dye at 5 ppm concentration up to three cycles. The composite membrane showed good stability and performance in the first two cycles; however, the removal efficiency significantly decreased in the third cycle. This reduction could be associated with membrane exhaustion due to an altered porosity or to the affected active sites of the nylon fibers, and integrated  $\alpha/\beta\text{-Bi}_2\text{O}_3$  particles that previously had sustained the capture of dye molecules and intense interaction with the generated radical species.

### 3.3 Antibacterial activity

**3.3.1 Bacterial inhibition in solid media.** The main aim of this study was to evaluate the potential of the composite membrane for the photocatalytic inactivation of waterborne pathogens *i.e.* *E. coli*. Therefore, initially, the inhibition tests were performed on *E. coli* cultured in solid media. The obtained results, showed in Fig. 8A, confirmed a growth inhibition halo against *E. coli* in the case of the composite membrane compared to the bare nylon membrane and control samples (biotic control). The calculated zone of

inhibition was around 14 mm. Moreover, under dark conditions, no significant inhibition zone was observed in the case of the composite membrane, revealing that the integrated  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  particles remained inactive and did not generate ROS against the growth of *E. coli*. Only a little zone of inhibition was recorded in some replicates, highlighting that probably the integrated  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  particles at high concentration can be a little bit toxic against bacterial growth, too. Meanwhile in the case of bare nylon and control samples (without membranes), a high growth of *E. coli* was observed in the whole Petri dish, both under dark and irradiated conditions, resulting in no significant effect on bacterial growth.

**3.3.2 Bacterial inhibition in a liquid suspension.** Once the antibacterial activity of the composite membrane was assessed in solid agar media, the photocatalytic response against a liquid stock solution of *E. coli* at a concentration of  $1 \times 10^6$  CFU  $\text{mL}^{-1}$  was evaluated by using the CPMR. The obtained results revealed a continuous *E. coli* reduction during the photocatalytic exposure *i.e.* as shown in Fig. 8B. The number of bacterial cells assessed by the plate count method of serially diluted samples collected from the CPMR at specified times confirmed the reduction of the cell density. It revealed 50% log reduction of *E. coli* after 90 min, and their complete removal after 240 min of treatment (as observed in Fig. 8B). To further confirm that the composite membrane has a biocidal effect on the microorganisms, fluorescence live/dead cell staining and imaging were applied *i.e.* using fluorescence microscopy. The samples were taken from the CPMR and stained, resulting in green-stained cells that represent living bacteria, whereas the red-stained cells are dead. The acquired fluorescence microscopy images of *E. coli* are shown in Fig. 9. In the beginning, plenty of live green-stained cells were observed, while at continued photocatalytic exposure, the red-stained cell increased their number. The obtained fluorescence images revealed that the bacterial cells were attacked by  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  in the composite membrane, resulting in the rupture of their cell membranes and allowing the PI dye to stain the DNA inside the nucleus,

as indicated from the observed red-stained cells. After treatment of 240 min, the higher proportion of red-stained cells over green-stained cells revealed that the  $\text{Bi}_2\text{O}_3$ /nylon composite membrane has not only a biostatic effect (growth inhibition) on the treated *E. coli* cells, but it acts as a biocidal material, able to kill this bacterial strain. During the photocatalytic reaction, the catalytic material integrated into the membrane could produce ROS that could attack the bacterial cells, lead to cell damage, and increase the permeability of the cell membrane towards PI staining dye.<sup>63</sup> Compared to these results, the bare nylon and control samples showed no significant red stained (dead) cells after prolonged exposure; the acquired fluorescence images for bare nylon treatment and control (without any membrane) samples are given in ESI† Fig. S4.

### 3.4 Photocatalytic mechanism

The photocatalytic particles partially embedded in polymeric/ceramic membranes remained active under irradiation and released reactive oxidative species (ROS) in water media and across the adsorbed/retained pollutants.<sup>61,64</sup> The dispersion of ROS across the membrane could be affected by mass-transfer limitations. However, due to the large surface created by the multi-layered nylon membrane and the exposed appearance of  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  particles on these layers, the large release of ROS could have been enough to reach the nearest adsorbed/retained pollutants, resulting in dye degradation and bacterial inactivation, compared to bare nylon. Indeed, the obtained results suggest that the  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  nanocomposite membrane has good adsorption assisted photocatalytic properties against the removal of both anionic and cationic dyes as well as bacteria. Some of the reported studies suggested ROS generation around the surface of heterojunction  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  (ref. 42) and nanocomposite materials *i.e.*  $\text{Sb}_2\text{WO}_6/\text{BiOBr}$  and  $\text{BiSbO}_4/\text{BiOBr}$ .<sup>38,39</sup> In the case of dyes, there could be various possibilities of their interaction with the generated ROS on the  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  active sites of the composite membrane. Either the dye molecules

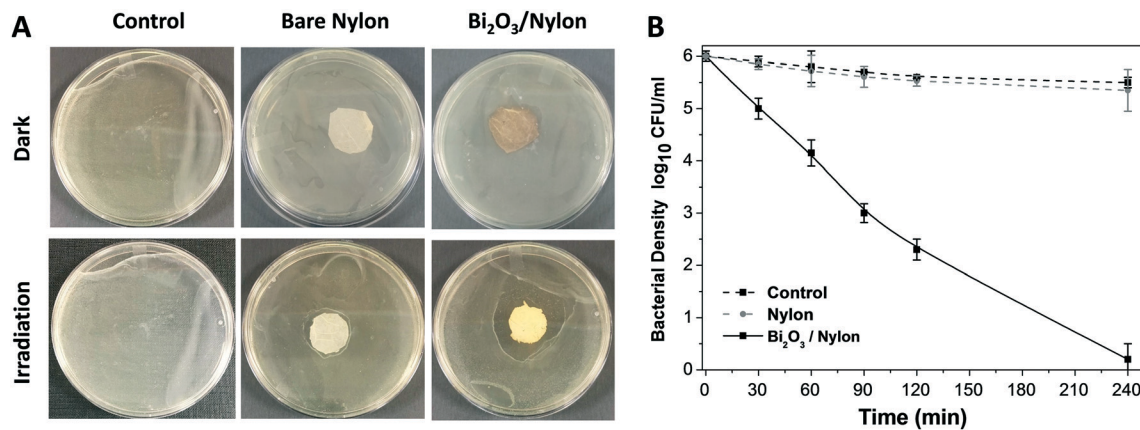


Fig. 8 A) Bacterial inhibition in solid media of control, bare nylon, and  $\text{Bi}_2\text{O}_3$ /nylon after incubation in the dark or under LED irradiation; B) *E. coli* density reduction in a liquid suspension treated with the photocatalytic  $\text{Bi}_2\text{O}_3$ /nylon membrane.

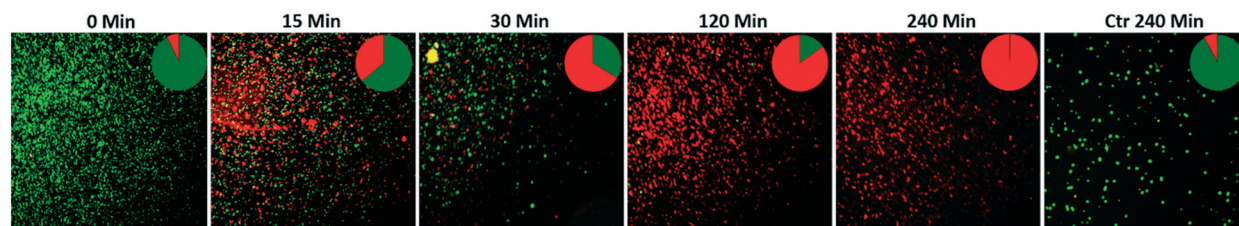


Fig. 9 Fluorescence stained images of live and dead *E. coli* cells at different treatment times.

of IC and RhB could have been adsorbed through ionic interaction *i.e.*, in the case of IC dye *via* sulphonic branched groups, while for RhB dye *via* the carboxylic group, or the dye molecules of IC and RhB were captured within the membrane pores of the multilayered nylon fibers. However, the good affinity of both dyes towards the composite membrane could be driven by both mechanisms *i.e.* through simultaneous ionic interaction and capture of dye molecules. After irradiation with visible light, the  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  particles in the composite membrane generated reactive species that attacked the captured/adsorbed dye molecules, performing adsorption-assisted photocatalysis. During the photo-discoloration of IC, the reactive species attacked the sulfonic groups, partially mineralized, and degraded into intermediate compounds identified as isatin sulfonic acid and 2-amine-5-sulfo-benzoic acid.<sup>65</sup> Meanwhile, the degradation of RhB followed the carboxylic acid oxidation and reduction path, being partly mineralized without any apparent shift of the absorbance in the UV-vis spectrum, typical of de-ethylation (see Fig. S3 in the ESI†).<sup>42</sup> In the case of the mixed solution, both dyes showed a good interaction and removal kinetics without any ionic inhibition/interference because of the equivalent interface provided by the membrane multilayers. Further, under prolonged exposure, the simultaneous degradation of both dyes occurred, due to the attack of ROS towards the IC and RhB dye molecules.

Concerning bacteria, the results obtained with *E. coli* revealed that probably the bacterial cells interacted with the composite  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  particles through multilayered channels, and were subsequently attacked by the generated ROS with resulting damage of the cell wall/membrane and their constituents. It is well-known that ROS can oxidise proteins, lipids and phospholipids,<sup>66</sup> damaging the membrane itself or simply its lipid bilayer organization, eventually leading to efflux of cytosolic contents or to cell lysis. These effects are particularly efficient on bacteria, fungi, and viruses, including multidrug-resistant strains, and due to their quick action, ROS are considered potent antimicrobial agents.<sup>67</sup> Similar mechanisms have been suggested for other antimicrobial metal and metal oxides, such as silver or copper, and are consistent with previous works that report the efficacy of  $\beta$ - $\text{Bi}_2\text{O}_3$  on *E. coli* and *S. aureus*.<sup>68,69</sup> The adsorption of the bacterial cells in the porous structure of the composite membrane combined with the action of the bismuth-based nanomaterial irradiated with LED visible light allowed a complete degradation of the pathogens. Indeed,

the ROS generated by the photocatalytic action of the  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  activated particles are more effective in close vicinity, due to their extremely low half-life. The increased number of red-stained (dead) cells at increased photocatalytic exposure showed that live cells interacted with and were attacked continuously by the photo-originated ROS. All of these results proved the potential applicability of bismuth-based composite membranes for water treatment and sanitation.

## 4. Conclusion

An  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$  powder semiconductor was synthesized by thermal annealing of  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  salt. The synthesized powder was integrated into the nylon-6 multilayered fibrous membrane *via* electrospraying, and characterized by XRD, FESEM, EDX, UV-vis, FT-IR and Raman analyses. The  $\text{Bi}_2\text{O}_3$ /nylon nanocomposite membrane showed a photocatalytic response against the separate solution of IC and RhB dyes, with a complete discoloration. The TOC analyses revealed the partial mineralization and degradation of IC and RhB treated solutions with the formation of stable and persistent intermediates. Further, in the case of the mixed solution of IC and RhB dyes, the composite membrane showed simultaneous degradation response to anionic IC dye and cationic RhB dye, due to their equal interaction with the reactive species, generated by  $\alpha/\beta$ - $\text{Bi}_2\text{O}_3$ . Moreover, dealing with a higher concentration of IC, the degradation kinetics was reduced by 3-fold, mainly due to the high saturation of the membrane layers by the dye-intermediates.

Furthermore, the composite membrane showed an antibacterial response against *E. coli* in solid and liquid cultures. In the case of solid *E. coli* media, the formation of a 14 mm inhibition zone was obtained, whereas, for the liquid *E. coli* suspension ( $1 \times 10^6$  CFU  $\text{mL}^{-1}$ ), the complete inhibition of bacterial growth was attained after 240 min of photocatalytic exposure. The induced bactericidal effect on *E. coli* was observed by fluorescence microscopy which confirmed the high proportion of dead over live cells. Further, the successive recycle tests employing the used membrane suggested the suitable applications and efficacy of the  $\text{Bi}_2\text{O}_3$ /nylon nanocomposite membrane for the removal of organic dyes and inactivation of waterborne pathogens.

## Conflicts of interest

There are no conflicts to declare.

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