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## Exciton-band tuning induced by the width of the cation in 2D lead iodide perovskite hybrids†

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It was previously demonstrated that the exciton energy of 2D hybrid organic–inorganic A<sub>2</sub>PbI<sub>4</sub> perovskites is correlated with the Pb–I–Pb bond angle of the inorganic layer. We demonstrate that this angle can be tuned by changing the width of the cation and that the exciton band energy can be varied over ca. 100 meV, at least within a limited series of A<sub>2</sub>PbI<sub>4</sub> compounds where A is a 2-(3,5-disubstitutedphenyl)ethylammonium or 2-(3-substitutedphenyl)ethylammonium ion and where the substituents are H, F, Cl, Br, or Me. Seven new crystal structures are presented here.

## Introduction

Recently, two-dimensional (2D) hybrid organic inorganic perovskites (HOIP) have been used as the active layer for solar cells<sup>1–10</sup> and light emitting diodes.<sup>11,12</sup> Compared with their 3D counterparts (such as MAPbI<sub>3</sub>, where MA = methylammonium), they can offer superior stability and a much wider structural diversity, since the cation size is not limited by the volume of the cuboctahedral cavity in the 3D inorganic lattice. The optical properties of the perovskite can be tuned by different methods, for example, by changing the connectivity of the octahedra<sup>13</sup> or the halide<sup>14–16</sup> used. The optical characteristics of these materials are particularly relevant to solar-cell and light-emitting diode applications. Previous work, both computational<sup>17,18</sup> and experimental,<sup>19</sup> on 3D lead iodide structures showed a relationship between the Pb–I–Pb bond angles and the onset of the optical absorption of the material.<sup>20–24</sup> On increasing the size of the cation (Cs<sup>+</sup> < MA<sup>+</sup> < FA<sup>+</sup>, where FA = formamidinium), the Pb–I–Pb angle decreases due to more distortion of the network of PbI<sub>6</sub> octahedra, resulting in poorer Pb–Hal orbital overlap and an increased band gap.<sup>25,26</sup> In 2D A<sub>2</sub>MX<sub>4</sub> (A = organic cation, M = metal, X = halide) HOIPs, there is considerably more pronounced excitonic character than in their 3D counterparts. However, trends in the exciton energy often closely follow those in the band gap. Moreover, in some instances, the exciton

energy has been found to also be correlated with the M–I–M angle.<sup>8,27–31</sup> The steric bulk of the cation has also been found to correlate with the Sn–I–Sn angle in some studies of A<sub>2</sub>SnI<sub>4</sub> perovskites.<sup>23,32,33</sup>

Knowing that the M–I–M angle correlates with the properties of the material, we were interested to probe whether the width of 2-arylethylammonium cations (defined as the distance between *meta* and *meta'* substituents) was related to this bond angle. We report the synthesis and crystal growth of a series of 2D perovskites where the aryl group have different widths; the *meta* substituents on phenylethylammonium (PEA) are systematically changed in the series (dY-PEA)<sub>2</sub>PbI<sub>4</sub> where Y = H, F, Cl, Br and Me (Fig. 1). We describe the structural and optical properties of these materials and compare them to those of other A<sub>2</sub>PbI<sub>4</sub> species reported in the literature.

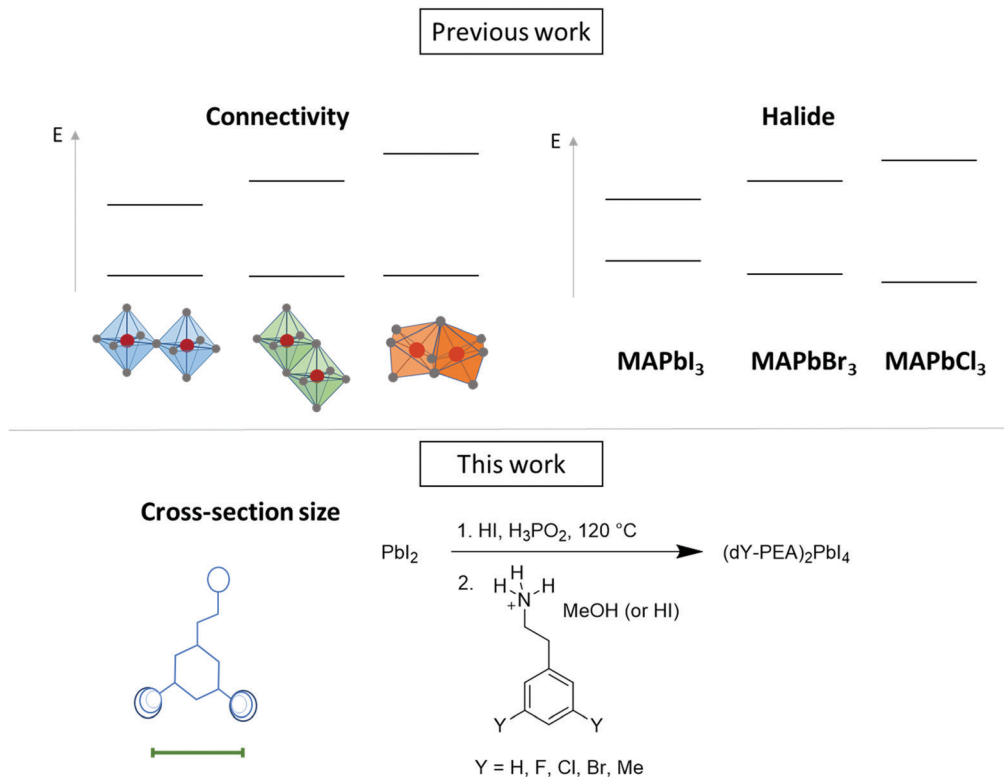
## Results and discussion

We investigated the structure and optical characteristics of lead iodide HOIPs with aromatic cations of increasing width. The width, defined here as the Y···Y distance, is incrementally enlarged by adding substituents at the 3,5-position of the PEA ring. The synthesis of the different cations is presented in the ESI.† The HOIPs were crystallized as pale to dark orange plates from a 1:2 solution of PbI<sub>2</sub> and A<sup>+</sup>I<sup>−</sup> in HI/MeOH (see ESI† for detailed synthesis). We abbreviate these HOIPs as PEA (2-phenylethylammonium), dF-PEA (2-(3,5-difluorophenyl)ethylammonium), dCl-PEA (2-(3,5-dichlorophenyl)ethylammonium), dBr-PEA (2-(3,5-dibromophenyl)ethylammonium), and dMe-PEA (2-(3,5-dimethylphenyl)ethylammonium). Of course, the increase of halogen atom size also results in an increase in polarizability, whereas dMe-PEA has approximately the same width as dBr-PEA, but is anticipated to be less polarizable. The purity of the crystals

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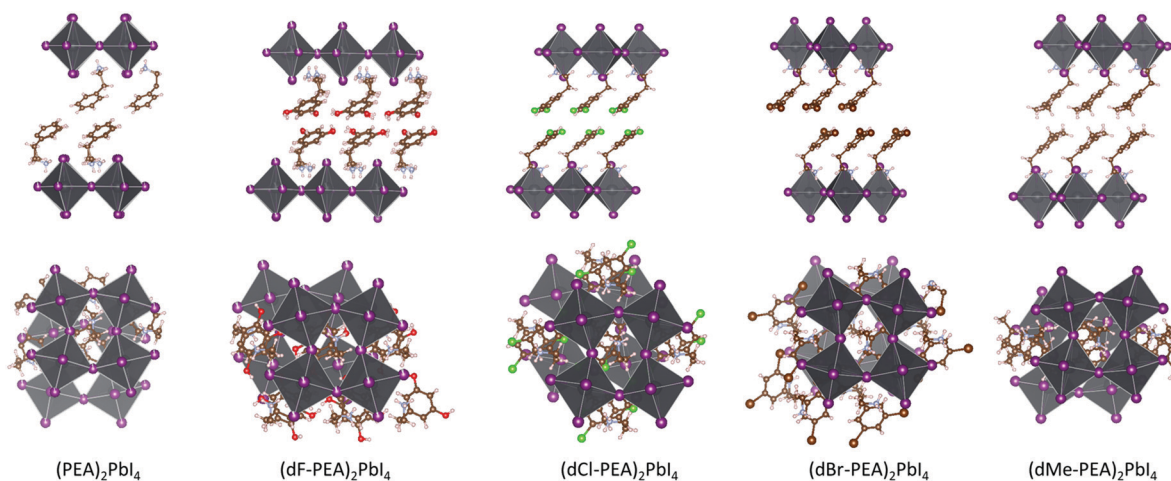


**Fig. 1** Top: Approaches to tuning the band energies of HOIPs. Trends in both excitonic and interband transitions of 2D and 3D HOIPs correlate well with the energy separation between valence and conduction bands within a series. Bottom: Approach presented here, where the width (atom–atom distance) of the cation is varied.

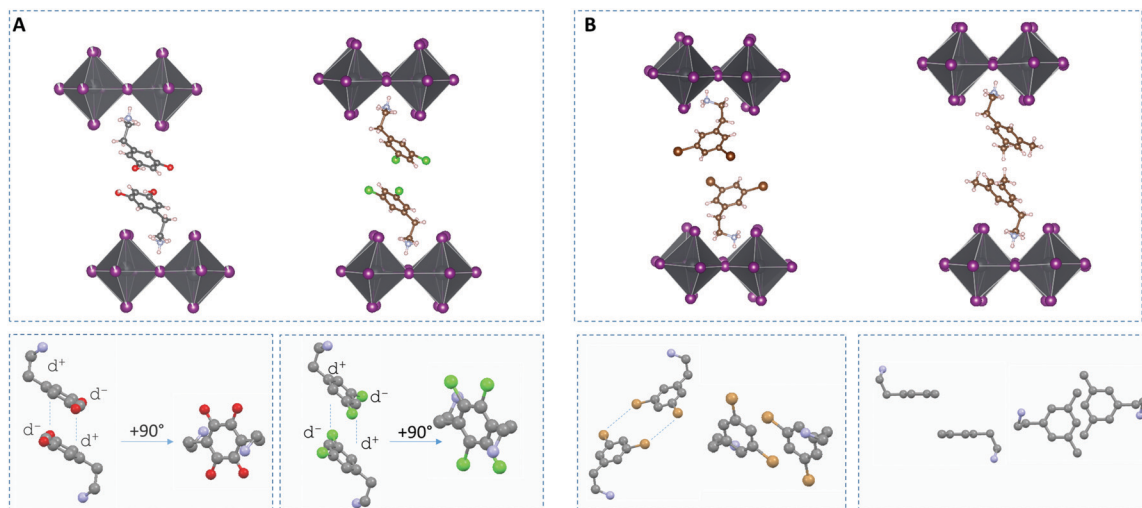
was assessed by elemental analysis and powder XRD; in each case, XRD patterns for powder samples were consistent with the corresponding single-crystal structures (see Fig. S1, ESI<sup>†</sup>; CCDC 1986784–1986789 and 1987208<sup>†</sup>). In all cases, we obtained  $\text{PbI}_4^{2-}$  layers of corner-sharing octahedra separated by bilayers of the organic cations (Fig. 2). In all compounds, there is hydrogen bonding between the iodides and the ammonium groups, which is typical for  $\text{A}_2\text{PbI}_4$  HOIP structures. Based on

the offset of successive inorganic layers, we can categorize dF-PEA, PEA, dCl-PEA and dMe-PEA as near-Ruddlesden–Popper structure and dBr-PEA as monocation-near-Dion–Jacobson structures (see offset in Table S3, ESI<sup>†</sup>).<sup>28</sup>

Thermogravimetric analysis of the four HOIPs based on 2-(3,5-disubstitutedphenyl)ethylammonium ions (see Fig. S2, ESI<sup>†</sup>) shows two distinct weight losses: one begins at *ca.* 230 °C, while our reference  $(\text{PEA})_2\text{PbI}_4$  starts exhibiting weight loss at 200 °C.<sup>12</sup>



**Fig. 2** Crystal structure of  $(\text{dY-PEA})_2\text{PbI}_4$ , Y = H, F, Cl, Br, Me, viewed along the plane of the inorganic sheet and after +90° rotation around the long in-layer axis. The structure of  $(\text{PEA})_2\text{PbI}_4$  is taken from the literature.<sup>34</sup>



**Fig. 3** Top: Partial crystal structure showing only two or three cations per four octahedra. Bottom: Cations showing the close approaches attributed to (A) dipole–dipole interaction ( $dY$ -PEA) $_2$ PbI $_4$ , where  $Y = F, Cl$  and (B)  $Br \cdots Br$  and  $CH_3 \cdots CH_3$  van der Waals interactions in ( $dBr$ -PEA) $_2$ PbI $_4$  ( $Br$ – $Br$  distance 4.15 Å) and ( $dMe$ -PEA) $_2$ PbI $_4$  ( $C$ – $C$  distance 3.94 Å).

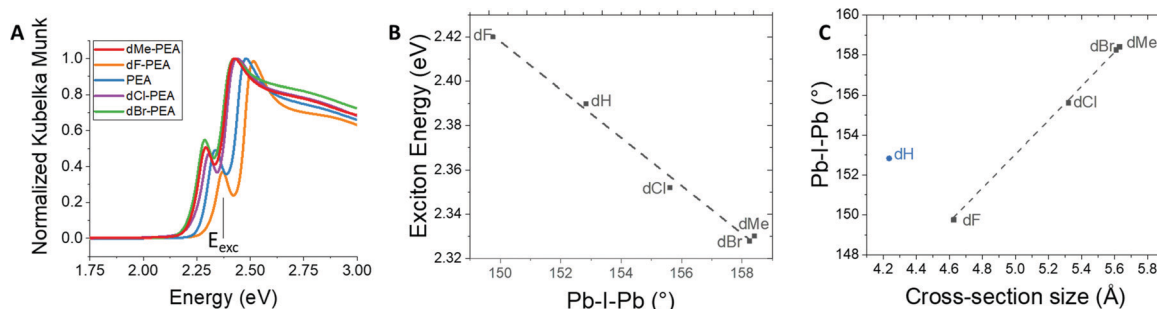
This initial weight loss is likely due to the organic cation decomposition and loss, while the second period of weight loss is presumably due to the sublimation of the lead iodide.

Multiple cation–cation interactions can be found in these HOIPs single crystals (see Fig. S3–S5, ESI†). For  $dF$ -PEA and  $dCl$ -PEA, the benzene rings from the two different organic layers are found to lay on top of each other. The distance between the centroid of one benzene to the plane made by another benzene is 3.343 Å for  $Y = F$  and 3.508 Å for  $Y = Cl$ , which could arise from a dipole–dipole interaction between the two cations (see Fig. 3A). In the case of  $dBr$ -PEA, there is a close approach between the bromine atoms from the two layers. This may be due to a  $Br \cdots Br$  van der Waals interaction (see Fig. 3B). A close approach between the methyl groups is also seen for  $dMe$ -PEA.

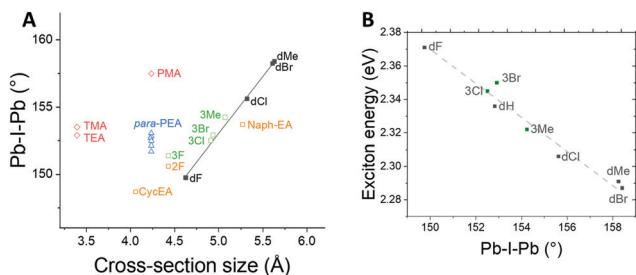
All compounds exhibited narrow linewidths for the photoluminescence feature characteristic of free exciton emission (see Fig. S5, ESI†). Fig. 4A shows the Kubelka–Munk transformation of the diffuse reflectance of the different HOIP powders and indicate the presence of two distinct bands. The energy

of the low-energy peak is assigned to the exciton band energy (see Fig. 4A). As expected, there is a linear relationship between the  $Pb$ – $I$ – $Pb$  angle and the exciton energy; the exciton band is red shifted by *ca.* 100 meV from  $dBr$ -PEA to  $dF$ -PEA (see Fig. 4B).<sup>27</sup> The relationships of  $Pb$ – $I$ – $Pb$  angle to the band gap in 3D HOIPs<sup>20–24</sup> and to the exciton-band energy in 2D HOIPs<sup>8,27–31</sup> have been observed in many instances as stated before. It has been shown that, in 3D HOIPs, band gap significantly opens upon octahedral tilting, mainly as a result of a decrease of the overlap integral, resulting in reducing antibonding character, and thus stabilization, of the valence band maximum.<sup>21</sup> Since trends in the exciton energy often closely follow those in the band gap, this rule can be extrapolated for the exciton energy of 2D HOIPs.

There is also a linear trend between the  $Pb$ – $I$ – $Pb$  angle and the width of the cation (see Fig. 4C). What emerges, based on these 5 compounds, is that, even if the reference compound (PEA, smallest cross-section size) does not follow the trend, we are able to show that the width of the cation is a *fair* indicator of the  $Pb$ – $I$ – $Pb$  angle in the HOIPs.



**Fig. 4** (A) Normalized Kubelka–Munk transform of diffuse reflectance data for the different HOIPs powders (with exciton energy,  $E_{exc}$ , indicated for the example of ( $dF$ -PEA) $_2$ PbI $_4$ ). (B) Relationship between the exciton energy and the  $Pb$ – $I$ – $Pb$  angle. (C) Relationship between the  $Pb$ – $I$ – $Pb$  bond angle and the width ( $Y$ – $Y$  distance calculated from the MM2-minimized geometry of the cation in Chem3D) ( $dH$  is excluded from the fit). The width of  $dMe$ -PEA is calculated as the average of the three longest H–H distances of the methyl.



**Fig. 5** Relationship between the Pb–I–Pb bond angle and the cross-section sizes (Y–Y distance calculated from the MM2-minimized geometry in Chem3D). The Pb–I–Pb bond angle of 2F-PEA, 3F-PEA, 4F-PEA, 4Cl-PEA, 4Br-PEA, 4Ph-PEA, TMA, TEA, NEA, CycEA, PMA were calculated using the reported crystallographic data. The red diamonds are compounds that don't follow the hypothesis. The blue (*para*-PEA), black (*dY*-PEA), green (3Y-PEA) and yellow (others) compounds represent different families of compounds and follow the hypothesis.

In order to establish whether this trend holds true for cations other than 3,5-disubstituted PEAs, we extended the study to other families of cations (see Fig. 5). First, we synthesized *meta*-substituted phenylethylammonium-based HOIPs (3Y-PEA, where Y = Cl, Br, Me) and determined the structures. Another member of this series, (3F-PEA)<sub>2</sub>PbI<sub>4</sub>, has previously been reported;<sup>35</sup> thus structural data are now available for both *dY*-PEA and 3Y-PEA families for the same set of substituents (Y = F, Cl, Br, Me). Again, the Pb–I–Pb bond angle increases with the width of the cation: the data points for the three new compounds essentially lie on the same line as those for the *dY*-PEA series, while the point for 3F-PEA compound still lies close to this line. Moreover, data for three newly synthesized compounds fit the same linear plot between the exciton energy and the Pb–I–Pb bond angle as the *dY*-PEA derivatives discussed above (see Fig. 5B).

We then examined additional structures available from literature reports. *para*-Substituted 2-phenylethylammonium-based perovskites (in which the cross-section size is constant) have similar Pb–I–Pb bond angles (A<sub>2</sub>PbI<sub>4</sub> where A = PEA 152.9°,<sup>34</sup> 4F-PEA 151.5°,<sup>36</sup> 4Cl-PEA 152.48°,<sup>36</sup> 4Br-PEA 152.15°,<sup>36</sup> 4Ph-PEA 153.1°,<sup>31</sup> 4Cl-Ph-CH(CH<sub>3</sub>)NH<sub>3</sub> 152.7°<sup>37</sup>), reinforcing the hypothesis that cross-section size is an important factor in determining the Pb–I–Pb bond angle, although the data points for these derivatives do not fall on the same line as the disubstituted and *meta*-substituted derivatives. A<sub>2</sub>PbI<sub>4</sub> perovskites for A = TMA<sup>38</sup> and TEA<sup>39</sup> cations (TMA = 2-(2-thienyl)methylammonium, TEA = 2-(2-thienyl)ethylammonium), which have the same cross-section size but different alkyl length, have similar Pb–I–Pb bond angles to one another, but larger than one would expect based on the trend in the disubstituted and *meta*-substituted compounds. On the other hand, Pb–I–Pb angle and cross-section size data for A<sub>2</sub>PbI<sub>4</sub> perovskites based on several other cations including 2-(naphth-2-yl)ethylammonium (NEA),<sup>34</sup> 2-(1-cyclohexenyl)ethylammonium (CycEA),<sup>40</sup> and 2-fluorophenylethylammonium (2F-PEA)<sup>41</sup> do lie close to the line defined by the (*dY*-PEA)<sub>2</sub>PbI<sub>4</sub> derivatives (and other (3Y-PEA)<sub>2</sub>PbI<sub>4</sub> derivatives) reported here.

There are cases of families of 2D A<sub>2</sub>PbI<sub>4</sub> structures reported in which the cross-section and Pb–I–Pb angle do not linearly correlate: (1) when the two layer of cations interpenetrate one another, instead of being separated in two sheets (*para*-substituted PMAs);<sup>28</sup> (2) when the 2D structure is not planar, but corrugated instead (as in the case of A = 2-(4-nitrophenyl)ethylammonium);<sup>42</sup> and (3) when the core of the cation is much larger than that of 2-(phenyl)ethylammonium (pyrene and perylene derivatives).<sup>5,43,44</sup>

## Conclusion

We report a total of 7 new 2D HOIPs containing 3,5-disubstituted-phenylethylammonium cations, *dY*-PEA (Y = F, Cl, Br, Me) and 3-substituted-phenylethylammonium cations, 3Y-PEA (Y = Cl, Br, Me). We were able to demonstrate that, at least for some limited series of compounds, the width of the cation correlates with the distortion of the inorganic sheet through Pb–I–Pb bond angle variation, which in turn correlates with the exciton energy. We note that while the second of these correlations has been discussed in other systems and can be understood in terms of effects on the perovskite band structure,<sup>21</sup> the first is purely an empirical observation. That is, the mechanism by which the cation width affects the bond angle is not clear, although, it is expected that changes in steric demands associated with variations of the cation width will affect the optimum packing in the interlayer region; these changes in packing may be accommodated by structural changes in both organic and inorganic portions of the structure. Given the complex energetic balance of interactions and the conformational flexibility of both the organic cations and the inorganic layers in these structures, the specific structural changes observed are not easily rationalized, as is underscored by the observed non-universality of the width/angle correlation. However, although width is not a universal factor determining the Pb–I–Pb angle and multiple other factors must likely be taken into account, the observation of this relation for at least some families of cations brings us closer to developing an understanding between the structure of the organic cation and the structure of the corresponding 2D HOIP. The deviations observed from the proposed relation may afford insight into other descriptors that could be looked at in the future.

## Conflicts of interest

There are no conflicts of interest to declare.

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