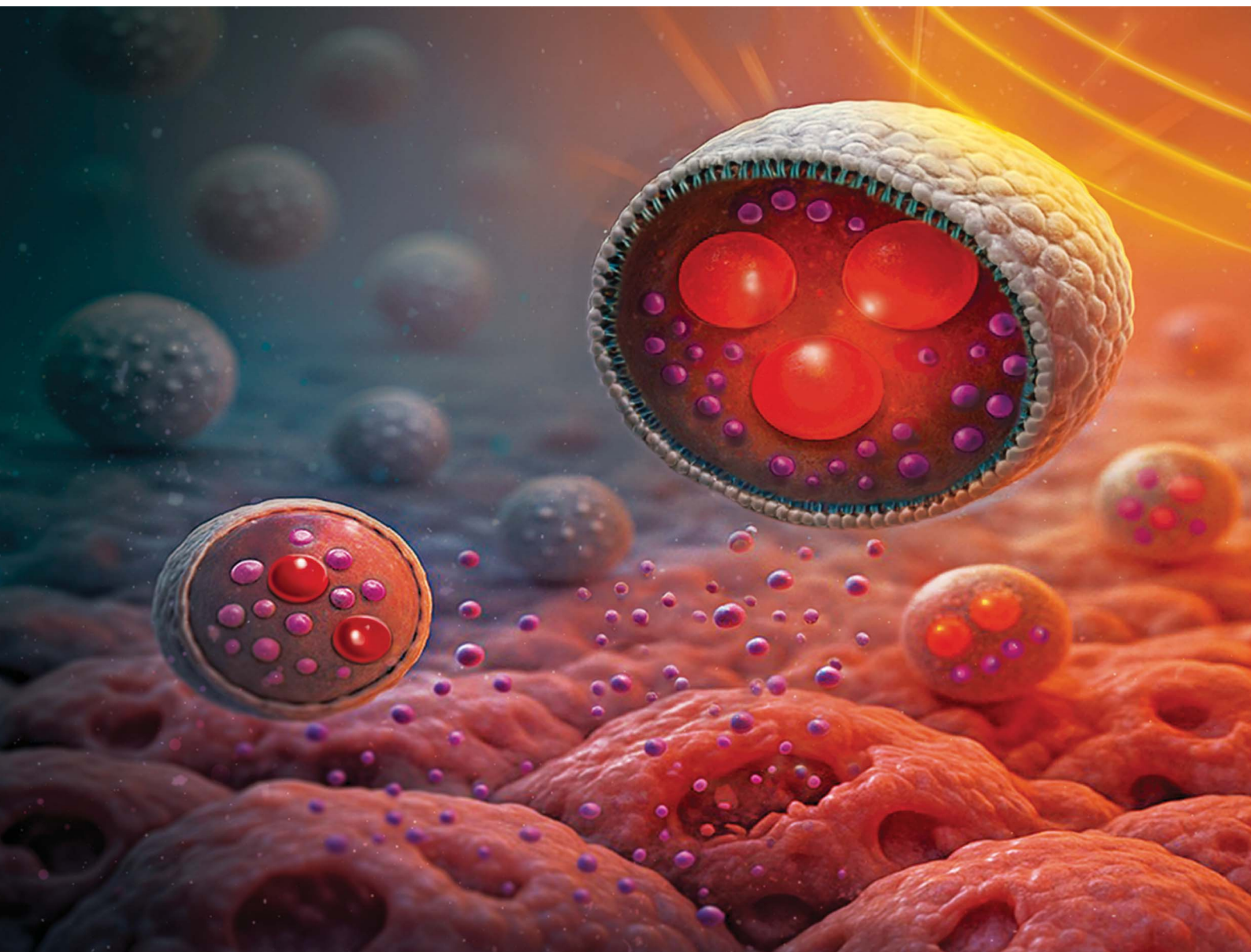


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Naveed Ahmed, Daniel Jaque Garcia, Mubashar Rehman *et al.*
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Magnetic hyperthermia-triggered multi-functional thermo-responsive lipid nanoparticles for enhanced paclitaxel release and cytotoxicity†

Muhammad Tayyab,^a Naveed Ahmed,^{*a} Muhammad Hisham Al Nasir,^b Ayesha Ihsan,^c Asim Ur Rehman,^a Khurram Shahzad Munawar,^d Daniel Jaque Garcia,^{*e} Maria del Carmen Iglesias de la Cruz^f and Mubashar Rehman^{*a}

The inadequate safety and efficacy of chemotherapy have led cancer medicine to focus on localizing drug delivery to the target. Thermoresponsive nanocarriers (liposomes and polymeric networks) exploit local hyperthermia to trigger targeted payload release; however, their low stability and unpredictable fate *in vivo* have led to failure in clinical studies. To overcome these challenges, we reported first-of-its-kind thermoresponsive lipid nanoparticles (TLNs) that undergo solid–liquid phase transition under hyperthermia to release the payload. This study enabled TLNs with on-demand drug delivery functionality to breast cancer cells by incorporating magnetically activated iron oxide nanoparticles (γ -Fe₂O₃) into a lauric and oleic acid-based phase-changing lipid-matrix to synthesize paclitaxel (PTX)- and γ -Fe₂O₃-loaded TLNs (P- γ -TLN). Critical independent variables were selected and then optimized using a central composite design to obtain the optimized formulation, P- γ -TLN 12, with a size of ~183 nm, polydispersity of 0.50, zeta potential of –22 mV, and encapsulation efficiencies of 85% for PTX and 60.49% for γ -Fe₂O₃. Thermoresponsive delivery was confirmed, with TLNs remaining relatively stable at 37 °C for 72 h, releasing only 34.26% of the drug, whereas exposure to 45 °C resulted in more than a two-fold increase, releasing 79.35% over the same period. Under an external alternating magnetic field, γ -Fe₂O₃ generated hyperthermia and induced a phase transition in P- γ -TLN, leading to abrupt drug release. Both γ -Fe₂O₃ and TLNs exhibited high biocompatibility, but TLN encapsulation significantly improved uptake in MCF-7 breast cancer cells. Under AMF, P- γ -TLN showed enhanced PTX release, resulting in more potent cytotoxicity against MCF-7 cells. The combination of high payload capacity, stimuli-responsive release, thermotherapy, and enhanced chemotherapeutic response highlights the substantial potential for TLNs in cancer therapy.

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1. Introduction

Breast cancer is the most prevalent malignant tumor, accounting for 23.8% of all cancers affecting women.¹ According to Breast Cancer Facts and Figures 2024–2025, an estimated 367 220 new cases of breast cancer in women and 2790 cases in men will be diagnosed this year in the United States alone.² There are several chemotherapeutic options for breast cancer; however, paclitaxel (PTX) is considered one of the most successful drugs to date. Clinically effective doses in cancer patients raise serious concerns due to its poor water solubility and associated toxicity. It has very low bioavailability due to poor permeability and solubility, classifying it as a biopharmaceutical classification system (BCS) Class IV drug.³ Chemotherapy and other traditional cancer treatments have not been entirely successful due to a lack of specificity, damaging both cancerous and healthy cells,⁴ as

^aDepartment of Pharmacy, Quaid I Azam University, Islamabad, 45320, Pakistan. E-mail: mtayyab@bs.qau.edu.pk; natanoli@qau.edu.pk; arehman@qau.edu.pk; mrehman@qau.edu.pk

^bDepartment of Physics, Riphah International University, Islamabad 44000, Pakistan. E-mail: hisham.nasir@riphah.edu.pk

^cNational Institute of Biotechnology and Genetic Engineering (NIBGE-C), Pakistan Institute of Engineering and Applied Sciences (PIEAS), Faisalabad, Pakistan. E-mail: aishaehsan@gmail.com

^dDepartment of Chemistry, University of Mianwali, 42200 Punjab, Pakistan. E-mail: khurramchemist@gmail.com

^eNanomaterials for Bioimaging Group (nanoBIG), Facultad de Ciencias, Departamento de Física de Materiales, Universidad Autónoma de Madrid, 28049, Spain. E-mail: daniel.jaque@uam.es

^fNanomaterials for Bioimaging Group (nanoBIG), Departamento de Fisiología, Facultad de Medicina, Universidad Autónoma de Madrid, Avda, Arzobispo Morcillo 2, Madrid, 28029, Spain. E-mail: mc.cruz@uam.es

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well as issues related to drug resistance, recurrence, and metastasis.⁵

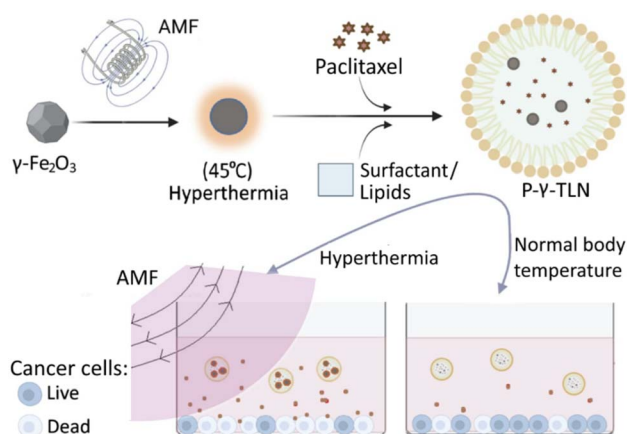
A notable shortcoming of clinically used nanoparticles is the limited spatial and temporal control of drug release. Most rely on passive release, which can result in either rapid drug leakage causing toxicity or slow release leading to reduced efficacy. On-demand release at the target site could revolutionize chemotherapy by minimizing side effects.⁶ The integration of magnetic hyperthermia treatment (MHT) into nanotechnology-based systems enables magnetically triggered heat generation under an alternating magnetic field.⁷ Various materials, such as lipids, micelles, polymers, and copolymers, have been employed for their thermoresponsive potential, yielding promising results.⁸

Similarly, low temperature-sensitive liposomes (LTSLs), such as ThermoDox®, aided by radiofrequency ablation (RFA), have reached clinical development (phase III) for augmenting stimuli-responsive drug availability in the hyperthermic (<40 °C) tumor microenvironment.⁹ However, liposomes involve complex synthesis methods employing toxic solvents,¹⁰ fusion, and drug leakage due to defects within lattice structures during storage.¹¹ Ultimately, the phase III clinical trial of ThermoDox® failed, *i.e.*, it did not increase progression-free survival (PFS) or overall survival (OS) due to long-term formulation instability and lack of supporting preclinical data.¹² Therefore, safe, effective, and feasible thermoresponsive nanoparticles for targeted chemotherapy of cancer were used in this research.

Limited intrinsic hyperthermia of cancer or loss of hyperthermia after partial treatment can compromise thermoresponsive function and lead to the development of lipid-based biomimetic magneto-vectors. These hybrids exhibit enhanced hyperthermia upon AMF exposure, ensuring complete payload delivery at the target site. Compared to polymeric systems, lipid-based nanoparticles demonstrate superior magnetothermal responsiveness, a potential that intrigues researchers in the field, with relatively higher drug loading and excellent biocompatibility.¹³ SLNs have been loaded in thermosetting hydrogels, where SLNs act as a drug reservoir and hydrogels as a thermoresponsive component.¹⁴ However, the gels are rather thermosetting, *i.e.*, forming a gel *in situ* at the normal body temperature rather than modifying drug release. The thermoresponsive lipid nanoparticles (TLN) were prepared from solid and liquid fatty acids for faster drug release upon exposure to hyperthermia, *i.e.*, <39 °C. The TLN undergoes a solid-liquid phase transition under hyperthermia, leading to faster drug release, higher cytotoxicity and uptake by cancer cells, and enhanced penetration across biological barriers.^{15,16} Later, different research groups have demonstrated the thermoresponsive drug release from lipid nanoparticles under different names.^{17–21}

In this study, we integrated iron oxide nanoparticles (γ -Fe₂O₃) in TLN to generate hyperthermia by externally applied AMF pulses, offering precise spatiotemporal control over thermoresponsive drug delivery. When loaded into TLN, PTX provided chemotherapy, whereas γ -Fe₂O₃ produced hyperthermia under AMF to induce drug release. All ingredients were

biocompatible and low-cost, and the preparation method was simple and feasible for industrial production.



2. Experimental

2.1 Materials

Lauric acid, oleic acid (OA), Span 80, Tween 80, Cremophor EL, and high-quality ethanol were purchased from Sigma-Aldrich (St Louis, USA). Paclitaxel (PTX) was generously gifted by Consolidated Chemical Laboratories (CCL) Ltd, Pakistan. Milli Q deionized water was used throughout the study. The dialysis bag (molecular weight 3.2 kDa) was purchased from Spectrum Labs (Rancho Dominguez, Canada). Ferrous chloride, ferric chloride, iron nitrate, ammonium hydroxide (33%), and nitric acid were all purchased from Merck, Germany.

2.2 Method of preparation and optimization

2.2.1 Thermoresponsive preoptimization. The thermoresponsive lipid mixture was tuned at a solid-liquid phase transition (melting point) of ~39 °C by mixing solid and liquid lipids per 100 mg of the total lipid mixture.¹⁶ Briefly, the liquid lipid and solid lipid were taken in a glass vial and heated to 10 °C above the melting point of lauric acid to reduce viscosity and ensure homogenous mixing. The melt was then taken in a capillary tube and allowed to settle down at room temperature to solidify. The melting point was then measured using the Stuart SMP20 digital melting point apparatus, ensuring the thoroughness of our examination.

2.2.2 Preparation of PTX and γ -Fe₂O₃ loaded TLN

2.2.2.1 Synthesis of maghemite (γ -Fe₂O₃). The γ -Fe₂O₃ nano-vectors, which were synthesized using the Massart method,²² exhibit unique properties such as high magnetization and biocompatibility, making them promising candidates for various applications in materials science and nanotechnology. Briefly, iron(II) chloride and iron(III) chloride were co-precipitated in an alkaline aqueous media at room temperature. To begin the synthesis, a molar ratio of 2 : 1 for Fe³⁺ : Fe²⁺ containing 0.04 M (5.40 g mol⁻¹) of iron chloride hexahydrate (FeCl₃·6H₂O) and 0.02 M (1.9854 g mol⁻¹) of iron chloride tetrahydrate (FeCl₂·4H₂O) were prepared in 25 mL of de-ionized water. The two solutions were mixed and slowly added dropwise into 250 mL of



de-ionized water under magnetic stirring. After 30 minutes of stirring, a 32% ammonia solution was gradually added at a rate of approximately 10–12 drops per minute. This addition was continued until the pH of the solution reached 10, resulting in the formation of brown precipitates. The solution was then stirred for an additional hour before allowing the precipitates to settle. The product of this co-precipitation was then washed several times and dried using the oven at 80 °C to obtain a powder of magnetite (Fe₃O₄) nanocrystals. The oxidation of Fe₃O₄ was carried out initially with nitric acid and then with Fe₃(NO₃)₃ to convert Fe₃O₄ into stable γ-Fe₂O₃ nanoparticles.²³

2.2.2.2 Synthesis of TLN. TLN was synthesized using a modified emulsification method.¹⁶ First, 18 mg of the OA was mixed with the PTX or/and γ-Fe₂O₃ (5–10 mg) in an acetonitrile solution and sonicated at 50 °C for complete dissolution and evaporation of organic solvent. This process yields coated inorganic nanoparticles that enhance hydrophobicity and colloidal stability.²⁴ 82 mg of lauric acid was then added and heated at 10 °C above the melting point to get a melted lipid mixture. Span 80 and Cremophor EL were added to the melted oil phase to enhance the homogenization and miscibility of the hydrophobic entities. The aqueous phase (2.3–5.6% of Tween 80), heated to the temperature of the oily phase, was added slowly and stirred until the microemulsion settled to room temperature. The final emulsion mixture was then centrifuged and pelletized. The obtained pellets were then redispersed in deionized water to obtain PTX-loaded TLN (P-TLN), γ-Fe₂O₃-loaded TLN (γ-TLN), and PTX and γ-Fe₂O₃ co-loaded TLN (P-γ-TLN).²⁵

2.2.3 Factorial optimization. This study optimized P-γ-TLN by a three-level, three-factor Central Composite Design-CCD (DoE, 12.0.3.0; State-Ease, Minneapolis). Independent variables were the amount of surfactant/co-surfactant (mg) (A), γ-Fe₂O₃ amount (mg) (B), and stirring speed (rpm) (C). The influence of these variables was then analyzed on the

corresponding dependent variables like particle size (Y1), zeta potential (Y2), PDI (Y3), EE of PTX (Y4), and EE of γ-Fe₂O₃ (Y5) (Table 1). A two-level factorial design was applied, augmented by star and central points for accurate estimation of the quadratic effects. The factorial design gave extreme values at +1 and −1, whereas circumscribed CCD pointed out each factor's new extreme values determined by the alpha value set at 1.68.²⁶ Levels were selected for every individual independent factor at surfactant/co-surfactant (low = 150 mg and high = 250 mg), γ-Fe₂O₃ (low = 5 mg and high = 10 mg), and stirring speed (low = 700 rpm and high = 1100 rpm).

This model, designed with axial points twice the number of factors, surfactant/co-surfactant (248.09 and 115.91), γ-Fe₂O₃ amount (3.29 and 11.70), and stirring speed (563.64 and 1236.36), was executed with utmost precision. The number of batches for this design was calculated using:

$$N = 2^k + 2k + nC, \quad (1)$$

where N is the number of formulation batches, K depicts the number of factors (3), n denotes the center points (6), and C is the number of times the experiments were repeated. Twenty runs (Table 1) were statistically tested against responses in terms of regression (analysis of variance) and graphical analyses (two-dimensional, *i.e.*, 2D contours plots and three-dimensional, *i.e.*, 3D response surface). Model-fitness was assessed by analyzing the values of coefficients of regression (R^2 , adjusted- R^2 , and predicted- R^2), F -values, standard error (SE), sum square error (SSE), coefficient of variance (% CV), mean, lack of fit, and adequate precision (ADP).²⁷

2.3 Characterizations

2.3.1 Determination of hydrodynamic diameter PDI and zeta potential. TLN, P-TLN, γ-Fe₂O₃, γ-TLN, and P-γ-TLN

Table 1 Experimental matrix with varying independent variables and respective responses for a set of runs in the designed space^a

Coding	A (mg)	B (mg)	C (rpm)	Y1 (nm)	Y2 (—)	Y3 (mV)	Y4 (%)	Y5 (%)
P-γ-TLN 1	200	7.5	900	255.8	0.58	−17.1	55.5	55.44
P-γ-TLN 2	284.1	7.5	900	197.7	0.57	−27.1	71	63.23
P-γ-TLN 3	250	10	700	318.3	0.50	−22.5	41.9	62.88
P-γ-TLN 4	200	7.5	1236	305.5	0.64	−20.9	68.5	60.27
P-γ-TLN 5	200	7.5	900	260.1	0.53	−19.9	57.3	55.34
P-γ-TLN 6	200	7.5	563	201.2	0.39	−16.8	47.2	47.44
P-γ-TLN 7	150	10	700	477.3	0.62	−16.6	33.5	56.55
P-γ-TLN 8	150	5	1100	291.9	0.76	−15.6	60.11	41.78
P-γ-TLN 9	200	7.5	900	259.6	0.56	−19.8	56.7	58.38
P-γ-TLN 10	250	10	1100	296.6	0.52	−25.3	61.76	68.34
P-γ-TLN 11	200	7.5	900	253.3	0.41	−17	52.5	55.39
P-γ-TLN 12	250	5	1100	183.1	0.51	−21.8	85.37	60.49
P-γ-TLN 13	150	5	700	329.5	0.42	−15	42.5	31.68
P-γ-TLN 14	200	7.5	900	261.3	0.59	−17.7	53.9	55.48
P-γ-TLN 15	150	10	1100	349.6	0.83	−16.6	38.27	61.11
P-γ-TLN 16	115.9	7.5	900	342.5	0.63	−14.8	40.1	46.75
P-γ-TLN 17	200	7.5	900	333.3	0.58	−17.4	54.5	63.11
P-γ-TLN 18	200	3.3	900	159.8	0.44	−16.7	72.5	42.11
P-γ-TLN 19	200	11.7	900	337.1	0.44	−21	44.5	63.34
P-γ-TLN 20	250	5	700	191.3	0.43	−21.1	67.5	57.22

^a A: surfactant/co-surfactant, B: γ-Fe₂O₃, and C: stirring speed; Y1: particle Size, Y2: PDI, Y3: zeta potential, Y4: EE of PTX, and Y5: EE of γ-Fe₂O₃.



dispersions were diluted 1 : 100 with de-ionized water and vortexed continuously until transparent. Photon correlation spectroscopy was employed using Zetasizer (Nano ZS, Malvern, Instruments, England) to evaluate the particle size (PS), zeta potential (ZP), and polydispersity index (PDI).²⁸

2.3.2 Morphological analyses. The γ -Fe₂O₃ and P- γ -TLN nanoparticles were analyzed using transmission electron microscopy (TEM; JEM1010, Jeol, 100 kV, CMOS 4k × 4k Tem-Cam F416 TVIPS, Gauting, Germany). All nanoparticle samples (diluted 1 : 10) were mounted on a copper grid and placed in a desiccator overnight for drying prior to imaging.²⁹

2.3.3 Encapsulation efficiency

2.3.3.1 Encapsulation of PTX. The encapsulation and loading efficiency (EE) of PTX in P-TLN and P- γ -TLN were determined using a modified protocol for high-pressure liquid chromatography (HPLC).³⁰ A combination of acetonitrile and water in a 70 : 30 ratio was utilized as a mobile phase along with 0.03% trifluoroacetic acid to run through the reversed phased column (C-18, 4.66 mm × 250 mm, 5 μ m) at 1 mL min⁻¹. The mobile phase pH was adjusted to 5.0 using glacial acetic acid, degassed for 60 min, and filtered using a 0.22 μ m nylon syringe filter. PTX was quantified by direct and indirect methods by measuring the absorbance at 227 nm using a calibration curve (0.025 to 2 μ g mL⁻¹). The chromatographic data obtained was processed using Lab Solutions software, and EE was calculated according to eqn (2).

$$\text{EE (\%)} = \frac{(\text{amount } (\mu\text{g}) \text{ of PTX in PTX loaded TLN})}{(\text{initial amount of PTX added})} \times 100 \quad (2)$$

2.3.3.2 Encapsulation of γ -Fe₂O₃. The untrapped γ -Fe₂O₃ in the colloidal dispersion was removed by gradient centrifugation to separate and could serve as a viable alternative to size exclusion chromatography (SEC) separations. Centrifugation (1000 rpm for 2 min) was carried out on P- γ -TLN to remove any excess of γ -Fe₂O₃, possibly not embedded in the lipid matrix,³¹ whereas entrapped γ -Fe₂O₃ was obtained by centrifugation (20 000 rpm for 15 min) until the supernatant is free from iron content. In an effort to achieve the highest purification of P- γ -TLN, SEC was employed to separate γ -Fe₂O₃. The preparations were poured onto a pretreated (soaked in PBS for 12 h at 30 °C, followed by 10 min centrifugation at 1000g) Sephadex G50 mini gel column. The separated P- γ -TLN were collected by a Retriever 500 fraction collector tube.³² To avoid the possibility of aggregated γ -Fe₂O₃ with larger size being co-eluted with the P- γ -TLN during the SEC procedure, the free γ -Fe₂O₃ were treated by the same procedure used for P- γ -TLN preparation without adding lipids and analyzed by SEC.

The iron content in purified P- γ -TLN was determined using a 55AA flame atomic absorption spectrophotometer (Agilent Technologies Inc., Santa Clara, CA) at 248.3 nm. Samples of P- γ -TLN were digested (1 mL suspension in a 5 mL acidic solution) in 6 M HCl overnight and subsequently diluted to the final concentration using 1% hydrochloric acid.³³ Standard solutions of γ -Fe₂O₃ were run in the 0.01–0.1 μ g mL⁻¹ range to establish a calibration curve after digesting them with concentrated HCl solution.

2.3.4 Compatibility analysis and solid-state characterization. The chemical interactions between the lipid matrix components and PTX/ γ -Fe₂O₃ were determined using an FT-IR spectrophotometer (IR Spirit-T, Diamond ATR, Shimadzu, Japan). The sample's spectra were recorded between 400 and 4000 cm⁻¹ with 2 cm⁻¹ scanning resolution. The recorded data, including the characteristic peaks of the lipid matrix, changes induced by the encapsulation of PTX/ γ -Fe₂O₃, and surface functionalization of γ -Fe₂O₃, holds significant importance in understanding these chemical interactions.³⁴

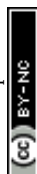
The determination of solid-state properties (crystalline or amorphous), physical compatibility, thermal change, and phase transitions were conducted using differential scanning calorimetry (DSC). The effect of the incorporation of PTX/ γ -Fe₂O₃ on the melting and crystallization temperatures of the formulated P- γ -TLN, transition temperature (*T_m*), and thermal endurance/stability were also recorded. Indium was used as a control with a heating rate of 10 °C min⁻¹ for PTX and γ -Fe₂O₃, whereas 1 °C min⁻¹ for lipid mixtures/TLN, variable temperature range of 25–350 °C, and nitrogen purge of 50 cm³ min⁻¹.³⁵

The XRD patterns of PTX, γ -Fe₂O₃, and P- γ -TLN were determined using Bruker D8-Advance diffractometer. All samples were analyzed in the 20–80° 2 θ range at a scan rate of 2° min⁻¹.³⁶

2.3.5 In vitro magnetic evaluations. The magnetization measurements were performed using a vibrating squid magnetometer (Lake Shore, 7404, Cryotronics, Westerville, Ohio, USA). The γ -Fe₂O₃ and γ -TLN samples were carefully dried, weighed, and sealed on a sample support to eliminate any potential magnetic interaction. The sample support was then placed in a tube that moved vertically in the zone of magnetic coils. The exact position of the sample support was determined using a scanning probe to precisely determine the sensitized zone of magnetic moments. The magnetic potential was finally measured as a function of changing the magnetic field between +50 kOe at 300 K, ensuring the highest level of accuracy.

The heating efficiency of the polydisperse magnetic materials was determined with a comprehensive approach, calculating the specific loss power (SLP) given by $\text{SLP} = A \times f$, where the area of the hysteresis loop is denoted by *A* and frequency by *f*. The value of SAR was calculated, determining the energy dissipation rate to the medium for the colloidal dispersion from the unit mass of solid magnetic material, leaving no room for doubt about the validity of the conclusions.

The magnetic heating efficiency of γ -Fe₂O₃ and γ -TLN was evaluated using nanoTherics Magnetherm NAN201003 (NanoTherics Ltd, UK) at various frequencies and magnetic field strengths: 163.8 kHz (*B* = 16.4 mT and 17 induction coil turns), 327.3 kHz (*B* = 17.1 mT and 17 induction coil turn), and 518.7 kHz (*B* = 23.9 mT and 09 induction coil turn). Chilled water was circulated to maintain the temperature of the field coils. The temperature was recorded for a certain time interval by a non-metallic fiber optic temperature sensor to avoid Eddy currents.³⁷ An amount of 1 mL of the Cremophor EL-based aqueous solutions (5 to 20 mg mL⁻¹) of γ -Fe₂O₃ and γ -TLN were taken in an Eppendorf tube with a hole in the cap to



submerge the temperature probe. Samples were placed in a thermally insulated polystyrene holder to ensure adiabatic conditions. The time-dependent temperature rise (ΔT) was measured for 400 s, while the heating efficiency (SLP) was calculated by utilizing the initial readings from the time-temperature rise curves using the following relation:

$$\text{SLP} = (C_{\text{samp}} \times \rho_{\text{samp}})/m_{\text{Fe}} \Delta T/\Delta t \quad (3)$$

C_{samp} is the specific heat capacity of the sample, ρ_{samp} is the density of the dispersion medium, m_{Fe} is the mass fraction of iron (in grams), and $\Delta T/\Delta t$ is the initial slope from the temperature-time curve. Moreover, intrinsic loss power (ILP, $\text{nH m}^2 \text{kg}^{-1}$, based on normalized SLP) values were also calculated per the following equation:³⁸

$$\text{ILP} = \text{SLP}/(H_0^2 f) \quad (4)$$

2.3.6 *In vitro* drug release study. The investigation of the thermoresponsive PTX release in P-TLN was conducted at 37, 41, and 45 °C using the dialysis bag method.³⁹ This involved dispersing suitable amounts of PTX and P-TLN in PBS (pH 7.4), enclosing them in a dialysis bag (molecular weight cut-off = 3.2 kDa), and suspending them in 100 mL of release medium at 150 rpm for 72 h. The PTX released into the dissolution medium was quantified using absorbance values by UV-vis spectroscopy at 235 nm. The percentage drug release was calculated using eqn (5):

$$\text{Drug release (\%)} = (\text{drug release concentration})/(\text{drug loading concentration}) \times 100 \quad (5)$$

To determine the magnetically triggered drug release at a field frequency of 518.7 kHz ($B = 23.8915$ mT and 09 induction coil turn), specific quantities of $\gamma\text{-Fe}_2\text{O}_3$ equivalent P- γ -TLN and P-TLN were exposed to a pulsative (ON and OFF) AMF for 10 min at pH 7.4. Briefly, 900 μL of the samples were poured into a dialysis bag of 3.2 kDa pore size, sealed tightly from both ends and then gently placed in a Falcon tube containing 50 mL of phosphate buffer. This assembly was then immersed in a water bath pre-set at 37 °C. After defined time intervals, the upper part of the tube containing submerged cellophane was exposed to an AMF-producing coil.⁴⁰ AMF pulses were applied to the P- γ -TLN and P-TLN to only increase in temperature up to 43 °C, a temperature that corresponds to the transition temperature of the TLN matrix coat. Samples were taken at variable intervals every 10 minutes upon exposure to AMF, *i.e.*, one before and one after the application, whereas OFF cycles were analyzed every 50 min. The drug release behavior was monitored by measuring the absorption of the drug using a UV-visible spectrophotometer.

2.3.7 Stability studies. The stability studies of P- γ -TLN were conducted under the intended time span at different storage conditions. At specified time intervals, samples were assessed for the particle size, PDI, drug, and $\gamma\text{-Fe}_2\text{O}_3$ content to assess the P- γ -TLN colloidal stability during storage conditions. The dispersions were labeled and stored after production in sealed glass vials at specific temperatures on the day of production and after every week for up to 12 months.⁴¹

2.3.8 Cell lines and cell culture. Human breast adenocarcinoma cell lines (MCF-7) were utilized for the study. MCF-7 was cultured in Dulbecco's modified Eagle medium (DMEM) along with 10% fetal bovine serum (FBS) and 1% penicillin-streptomycin. The 2×10^4 cells were seeded in 96-well plates, placed in an incubator at 37 °C, 5% CO_2 , and tested after reaching 90% confluency.⁴²

2.3.8.1 *In vitro* cytotoxicity. The dimethyl thiazolyl diphenyl tetrazolium (MTT) colorimetric assay was conducted to evaluate the cytotoxic activity of the free PTX, $\gamma\text{-Fe}_2\text{O}_3$, Cremophor EL, blank TLN, P-TLN, and P- γ -TLN against cancerous (MCF-7) cells. The cells were incubated with the different formulations under study for 48 h. After incubation, the cells were washed with PBS buffer and 100 μL of fresh complete medium containing MTT (10%, 5 mg mL^{-1}) and subsequently incubated for another 4 h. Finally, MTT was removed, and DMSO was added to dissolve MTT crystals for another 10 min at 37 °C. Cells treated with PBS were considered as negative control, whereas the ones treated with Triton X 100 were positive control. The absorbance of the free MTT was calculated after measuring absorbance on the microplate reader (BioTek-ELx800, USA).⁴³ The cell viability percentage was calculated based on eqn (6).

$$\text{Cell viability (\%)} = (\text{absorbance of treated cells})/(\text{absorbance of control cells}) \times 100\% \quad (6)$$

2.3.8.2 *In vitro* chemo-magnetic hyperthermic cytotoxicity. Cytotoxicity analysis of $\gamma\text{-Fe}_2\text{O}_3$, γ -TLN, and P- γ -TLN with or without AMF treatment, was carried out at a concentration of 500 $\mu\text{g mL}^{-1}$. For the AMF-triggered chemo-hyperthermia therapy, the cells were incubated with plain and lipid-embedded $\gamma\text{-Fe}_2\text{O}_3$ and were exposed to AMF of 518.7 kHz ($B = 23.9$ mT and 09 induction coil turns) for 10 min. The cell viability was assessed using the MTT assay after the required incubation time with 100 μL MTT solution as described previously.⁴⁴

2.3.8.3 *In vitro* cell uptake imaging. A qualitative cellular uptake study was conducted with MCF-7 (1×10^5 cells), seeded and kept for adherence overnight. The culture medium was decanted and washed with 200 μL of PBS to remove the serum of the culture medium to avoid any interference in the cellular uptake of the dye or nanoparticles. The cultured cells were then treated with a 50 μM concentration of 100 μL of rhodamine-B or an equivalent concentration of rhodamine-B-NPs and incubated for 45 min at 37 °C in a 5% CO_2 incubator. At the end of the incubation, the cells were washed thrice with 200 μL PBS to remove the adsorbed dye or nanoparticles. The cells were stained with DAPI (4',6-diamidino-2-phenylindole-1 $\mu\text{g mL}^{-1}$) for 10 min, washed three times with PBS and then visualized under a fluorescence microscope (Nikon Eclipse 2 Ti2 with light emitting diode lamp-pE-300 light series, CoolLED), focusing light on the sample using a 40 $\times$, 0.35 NA objective lens (SLMPLN, Olympus). Images were acquired with a metal oxide semiconductor camera (sCMOS) (OrcaFlash4.0, Hamamatsu), and a standard series DAPI filter cube and GLP filter cube from Nikon were employed.⁴⁵



3. Results and discussions

3.1 Method of preparation and optimization

3.1.1 Thermoresponsive preoptimization. The melting point of lauric acid, a 12-carbon saturated fatty acid, was first determined (48 °C). This was then blended with a liquid lipid OA (18-carbon monounsaturated) to solubilize the maximum amount of the drug and $\gamma\text{-Fe}_2\text{O}_3$. The quantity of liquid lipid was adjusted to tune the melting point of the lipid mixture more than the room temperature, *i.e.*, solid-state at 25 °C.⁴⁶ Our research on the solid lipid/liquid lipid mixtures and their melting points as a function of solid lipid proportion in the overall lipid mixture has practical implications. By applying a straight-line equation, the precise lipid mixture composition for thermoresponsive targeting was calculated. The resulting standard curve showed a clear decrease in the melting point with an increase in liquid lipid, *i.e.*, a decreasing solid lipid proportion in an overall lipid blend. This finding suggests that by adjusting the lipid composition, the melting point can be controlled, which could be beneficial for various applications.⁴⁷ The desired lipid mixture with the thermoresponsive melting point was successfully achieved using the straight-line, $y = mx + c$ or $y = 0.472x + 1.75$ (y -melting point, m -slope, and c - y intercept). Substituting the value of the thermoresponsive point of 39.0 °C required a solid lipid proportion of 78.91 mg, which was further confirmed with DSC analysis.

3.1.2 Preparation of TLN. Fe_3O_4 is oxidized to either $\gamma\text{-Fe}_2\text{O}_3$ or $\alpha\text{-Fe}_2\text{O}_3$ (hematite) depending upon the particle size of Fe_3O_4 and the temperature of the reaction mixture. The nanoparticles of $\gamma\text{-Fe}_2\text{O}_3$ and Fe_3O_4 , with their unique and valuable properties, have more magnetic potential than hematite when compared to saturation magnetization levels. This oxidation process, influenced by the particle size and temperature of the reaction mixture, is a key area of research. For instance, Fe_3O_4 below 3000 Å continually oxidizes to $\gamma\text{-Fe}_2\text{O}_3$ when treated below 220 °C, while Fe_3O_4 larger than 3000 Å will first yield intermediate ferric oxyhydroxides,⁴⁸ which will later be converted to hematite. The $\gamma\text{-Fe}_2\text{O}_3$ conversion was carried out by ferric nitrate solution at 100 °C for enhanced magnetic properties and chemical stability and to avoid high-temperature hematite conversion at 250 °C. This temperature control was achieved using a lab-made reactor of an inverted bell jar plugged with cotton and placed on a hot plate with a temperature probe to continuously monitor the temperature of the vessel.

Our study stabilized the drug delivery system with two non-ionic surfactants: relatively minor sorbitan mono-oleate (Span 80) and larger polyoxyethylene sorbitan mono-oleate (Tween 80). Tween 80 is an aqueous, non-ionic surfactant with an HLB value of 15, whereas Span 80 is a hydrophobic surfactant with an HLB value of 4.3. A combination of non-ionic surfactants, Tween 80 and Span 80, were used to form a compact surfactant coating and prevent particle coalescence. The long-chain alkyl groups in both emulsifiers prevented polymorphic transitions, thus resulting in more stable TLN formation.⁴⁹ Furthermore, they are more biocompatible than cationic, anionic, or amphoteric counterparts.⁵⁰

A modified emulsification method produced four formulations (TLN, P-TLN, γ -TLN, and P- γ -TLN). TLN and P-TLN appeared as milky white homogenous emulsions, while γ -TLN and P- γ -TLN appeared as milky orange emulsions due to the iron oxide contents.

3.1.3 Factorial optimization. Design expert® proposed and analyzed twenty experimental runs for the central composite design statistically. The correlation between the two sets of variables was put into different mathematical functions to check the best possible for the available design space. These different model designs were linear, 2FI (two-factor interaction), cubic, and quadratic. The best-fit model was validated by examining the values of the analysis of variance (ANOVA), F -value, lack of fit P -value, coefficient of determination (R^2), adjusted- R^2 , predicted- R^2 , coefficient of variance (% CV), and signal-to-noise ratio by measuring adequate precision (ADP). Model parameter (Table S1†) with P -value of <0.05 , lack of fit P -value >0.05 , large F -value, high R^2 , difference between R^2 (adjusted and predicted) less than 0.2, % CV of less than 10%, and value of ADP >4.0 are indicative of model reliability and adequacy.⁵¹ Particle size, polydispersity, and zeta potential are correlated to independent variables following the linear model, whereas entrapment of PTX and $\gamma\text{-Fe}_2\text{O}_3$ followed 2FI.

Interrelation between several variables was assessed using polynomial equations.

$$\text{Particle size} = +280.24 - 51.44A + 54.49B - 1.45C \quad (7)$$

$$\text{Polydispersity} = +0.5496 - 0.0581A + 0.0269B + 0.0793C \quad (8)$$

$$\text{Zeta potential} = +19.03 + 3.84A + 1.08B + 0.8051C \quad (9)$$

$$\begin{aligned} \text{EE of PTX} = & +55.26 + 9.82A - 9.31B + 7.02C \\ & - 2.30AB + 1.92AC - 1.36BC \end{aligned} \quad (10)$$

$$\begin{aligned} \text{EE of } \gamma\text{-Fe}_2\text{O}_3 = & +55.32 + 6.26A + 6.84B + 3.29C \\ & - 3.84AB - 0.7413AC - 0.4187BC \end{aligned} \quad (11)$$

Positive and negative signs before a variable in a polynomial equation determine the synergistic and antagonistic effects, respectively.⁵² Equally important, the magnitude of the response is defined by the coefficients of the independent variables. Understanding these coefficients is a key aspect of interpreting polynomial equations. The interconnection of independent *vs.* dependent variables was further elucidated by 3D surface and 2D contour plots. Sensitivity and insensitivity towards the response variable were well illustrated by the steepness and flatness of the curve, respectively.⁵³

The experimental runs yielded particle sizes ranging from 159.8 to 477.3 nm (Table 1). The ANOVA results revealed both significant and insignificant effects, with the stirring speed showing a significant effect in a sequence of $C > 0.05$ and $B > A$ ($0.0002 < 0.0004$), respectively (Table S2†). Fig. 1-1A demonstrates a linear increase in particle size with the rise in $\gamma\text{-Fe}_2\text{O}_3$ amount, while the surfactant/co-surfactant had a significant decreasing effect, and stirring had an insignificant effect (0.9018). The polynomial equation for particle size showed a strong positive correlation between the $\gamma\text{-Fe}_2\text{O}_3$ amount and the nanoparticle



size.⁵⁴ This effect, depicted in Fig. 1-1A, could be explained in terms of the lipid-to-aqueous ratio, a crucial factor for the nanoparticle size produced by lipid dispersion. The size of the nanoparticles decreased by decreasing the amount of solid lipid due to the change in the system's viscosity. A less viscous lipid mixture will bear more homogenization, and strong shear will result in smaller nanoparticles. At the same time, a more viscous system will resist the movement of fluid⁵⁵ and will not break down into smaller particles, which is valid for the increased amount of lipid/drug/ γ -Fe₂O₃. The surfactant/co-surfactant amount and stirring had an inverse relationship, *i.e.*, a decreasing particle size when increasing both (Fig. 1-1C). A system with higher viscosity produces bigger-sized particles; the addition of surfactant reduces the interfacial tension more effectively, thereby reducing particle sizes.⁵⁶ Likewise, stirring produces the desired shear to break the matrix to fine rearrangement, making small-sized nanoparticles, but the coefficient (−1.45) represents a very meagre synergistic effect (Fig. 1-1B and 1C).

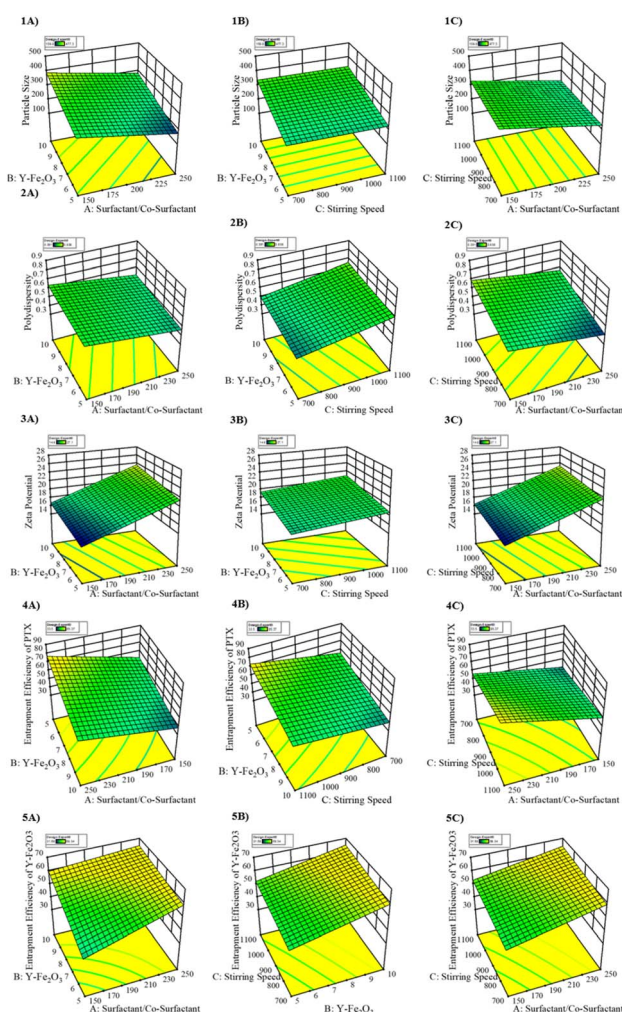


Fig. 1 3D graphical interpretation of the interaction factors (A) AB, (B) BC, and (C) AC at the mid-level of C, A, and B, respectively, for particle size (1A–C), polydispersity (2A–C), zeta potential (3A–C), entrapment efficiency of PTX (4A–C), and entrapment efficiency of γ -Fe₂O₃ (5A–C).

The variability in polydispersity with different alterations in variable quantity depicted a linear decrease in increasing the surfactant/co-surfactant and stirring speed (Fig. 1-2C). The amount of payload increased the polydispersity from 0.426 (5 mg) to 0.838 (10 mg) for formulations P- γ -TLN 13 and P- γ -TLN 15, respectively. ANOVA depicted a maximal response by stirring, whereas the payload had an insignificant response ($C > A > B$). At a higher level of surfactant/co-surfactant concentration, the surfaces of the TLN were covered effectively, which prevented them from aggregating and resulting in a small polydispersity index.⁵⁷ Surfactants lessen the contact angle between the lipid matrix and the aqueous phase, causing less particle aggregation.⁵⁸ Fig. 1-2A depicts the higher concentration of γ -Fe₂O₃ caused by untrapped particles in the suspension, resulting in increased chances of agglomerations.⁵⁹

The zeta potential results were subjected to ANOVA, a powerful tool that depicted the significant role of all independent variables in a sequence of $A > B > C$ with *P*-values in the increasing order of $<0.0001 < 0.0073 < 0.0359$, respectively. This statistical analysis provided a comprehensive understanding of the impact of each variable on the zeta potential. Anionic carboxylate headgroups in lauric acid render an overall negative charge,⁶⁰ which is presented on the nanoparticle's surface. A zeta potential of around 30 mV is an important determinant, predicting the stability of colloidal formulations,⁶¹ preferentially by steric stabilization and electrostatic repulsion. All formulations had a negative charge even though the particles were stabilized using two non-ionic surfactants, Tween 80 and Span 80. This negative charge was attributed to the solid lipid and preferential adsorption of OH[−] ions from water by lipid particles. The surface of the nanoparticles was stabilized by the non-ionic emulsifiers, but they had an overall negative charge. This charge was attributed to the negatively charged fatty acid chains of the solid lipid and OH[−] ions adsorbed by the lipid matrix from the water molecules. Additionally, the polyoxyethylene (POE) dense tails extending to the lipid matrix provided steric hindrance and colloidal integrity.⁵⁶ The non-ionic surfactant decreased the zeta potential due to the shielding of endogenous negative charge of the lipid matrix.⁶²

The ANOVA analysis depicted that EE of PTX/ γ -Fe₂O₃ was remarkably changed ($P < 0.05$) with the alteration in all the individual independent variables (<0.0001). This means that the efficiency of PTX/ γ -Fe₂O₃ is significantly affected by changes in

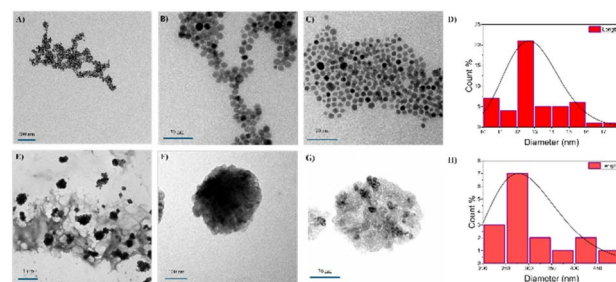


Fig. 2 Morphological analysis of γ -Fe₂O₃ (A)–(C) and P- γ -TLN (E)–(G) at various resolutions. Size distribution analysis using ImageJ for γ -Fe₂O₃ (D) and P- γ -TLN (H).



the independent variables. In contrast, combination factors had only a significant effect on *AB*, whilst others had an insignificant role in altering the surface potential (*AC* and *BC*). Surfactant/co-surfactant and amount of PTX/ γ -Fe₂O₃ increased the drug loading/entrapment efficiency and stabilized the system (Fig. 1-4A-C and 5A-C). The increased initial payload made more lipophilic drugs available to the lipid matrix, increasing the partitioning and drug entrapment.⁶³ Surfactant/co-surfactant also decreased the matrix's crystallinity and improved the payload's solubilization to improve entrapment. Similarly, hydrophilic oxyethylene groups in Tween 80 furnished steric hindrance to prevent coalescence and aggregation of nanoparticles. The role of Span 80 was to anchor lipophilic entities like PTX and γ -Fe₂O₃ in the lipid matrix to prevent their leakage and diffusion into the aqueous phase.⁶⁴ The point-prediction function was used to select an optimized formulation based on employing different sets of goals (minimize particle size, minimize polydispersity, maximize zeta potential, maximize payloads) for all the responses with the desirability of 0.793. The selected formulation (P- γ -TLN 12) had 250 mg surfactant/co-surfactant (*A*), 5 mg γ -Fe₂O₃ (*B*), and 1068 stirring speed (*C*). The experimental optimized batch with the projected independent variables was developed and re-characterized for the response variables. The projected responses (particle size-173.08 nm, polydispersity-0.531, zeta potential-22.12, EE of PTX-85.37%, and EE of γ -Fe₂O₃-61.08%) and the obtained result (particle size-183.1 nm, polydispersity-0.507, zeta potential-21.8, EE of PTX-85.37%, and EE of γ -Fe₂O₃-60.49) were in a reasonable agreement, representing the optimized formulation's sanity.

3.2 Characterizations

3.2.1 Determination of hydrodynamic diameter PDI and zeta potential. The average particle size of γ -Fe₂O₃ and TLN was 17.06 + 2.23 nm and 145.5 + 6.6 nm, whereas that of the optimized formulation P- γ -TLN 12 was 183.1 + 10.24 nm, slightly higher than the unloaded nanoparticles (Fig. S4†). The particle size was affected mainly by the amount of payload and surfactant/co-surfactant due to increased solubilization of the payload in the lipid matrix. The surface charge of P- γ -TLN fluctuated between -14.8 and -27.1 mV. In contrast, blank TLN had -19.3 mV, depicting it relied mainly on the solid lipid available in the matrix and the non-ionic surfactants coated on the surface of nanoparticles. The γ -Fe₂O₃ had a zeta potential of +1.64 mV, causing poor polydispersity and agglomeration, which was surface functionalized to improve the colloidal stability. Polydispersity for optimized P- γ -TLN 12 was 0.507, implying the formulation is polydisperse⁶⁵ due to a moderate zeta potential value. Given the higher shear required to completely homogenize the oleic-coated γ -Fe₂O₃, the oily phase is more viscous and difficult to disperse. Sonication and stirring tend to make agglomerates of the γ -Fe₂O₃ particles, thus increasing the size and PDI in all the formulations.⁶⁶ This aggregation also tends to clog the syringe filters, hindering the filtration process before the particle size analysis. Similar results were obtained by Jose *et al.* and Shimojo *et al.* for the

NLCs prepared with high shear homogenization with PDI of larger than 0.5, demonstrating the process of emulsification was the reason for creating aggregates.^{67,68}

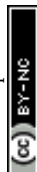
3.2.2 Morphological analysis. TEM analysis illustrated that the γ -Fe₂O₃ nanoparticles had spherical surfaces with no artifacts (Fig. 2A-C), whereas P- γ -TLN had irregular surfaces due to the presence of γ -Fe₂O₃ in the lipid matrix. The size obtained was approximately 10 nm for γ -Fe₂O₃, which corresponds to the hydrodynamic sizes obtained by the DLS with a slight difference due to the shrinkage of the solvation layer (γ -Fe₂O₃)⁶⁹ and 300 nm for P- γ -TLN due to the polydispersity in particles. Size distribution was performed using Image J, which revealed the γ -Fe₂O₃ was monodisperse owing to its sharp distribution curve (Fig. 2D), while the P- γ -TLN has a broad distribution curve (Fig. 2H), depicting it to be polydisperse complying with the DLS measurements. The average particle size obtained for γ -Fe₂O₃ was 12.963 + 0.123 nm (min-10.338 and max-17.646 nm for 50 particles count), whereas P- γ -TLN had 306.625 + 0.258 nm (min-202 and max-472 nm for 15 particles count).

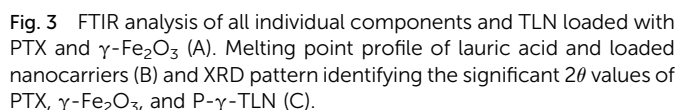
3.2.3 Encapsulation efficiency

3.2.3.1 Encapsulation of PTX. The quantitative detection of PTX in P- γ -TLN was carried out on HPLC equipped with a UV detector. PTX was eluted by employing ACN and aqueous 0.03% TFA mixtures at a pH of 5 without a buffer system and a short retention time of 4.55 min. The main advantage of this method was the column safety and short-time analysis. The linearity of the method was established by running standard solutions at five different concentrations from 4.74 to 62.86 μ g mL⁻¹, demonstrating the precision of our approach. The accuracy was ensured using a recovery test by direct and indirect methods fortifying with the PTX standard at three levels, confirming there is no possible overlap. The amount of drug used to prepare TLN was 5 mg, and the mean entrapment values for both methods for P- γ -TLN 12 were 85.37 + 3.44 and 84.89 + 3.55, respectively.

3.2.3.2 Encapsulation of γ -Fe₂O₃. In P- γ -TLN, a fatty acid-based solid lipid formed a solid-phase matrix with an inner core containing γ -Fe₂O₃. The incorporation of γ -Fe₂O₃ depends on the number of particles that can be accommodated.⁷⁰ For this reason, gradient centrifugation in conjunction with atomic absorption spectroscopy (AAS) was used to determine the untrapped and entrapped iron content. The entrapment for all the batches was between 31.68% and 68.34%, with the optimized batch P- γ -TLN 12 having EE of up to 60.49%. This untrapped iron content was also separated, with no significant difference between either of the techniques. These findings demonstrated the capability of TLN to not only load small molecular drugs with high efficiency but also achieve remarkable entrapment of colloidal entities, such as γ -Fe₂O₃. The EE of P- γ -TLN and PTX is sufficient for cancer treatment, instilling confidence for further testing.

3.2.4 Compatibility analysis and solid-state characterization. The characteristic peak for lauric acid at 2847, 2915, and 2953 cm⁻¹ was attributed to the C-H stretching vibration,³⁰ whereas carbonyl peaks were observed at 1695 and 1471 cm⁻¹ (Fig. 3A).⁷¹ In OA, the C=O stretch was illustrated as an intense peak at 1708 cm⁻¹ (ref. 72) while two other sharp bands appeared at 2853 and 2922 cm⁻¹ due to the symmetric CH₂





In the oleic acid-coated $\gamma\text{-Fe}_2\text{O}_3$, the C=O stretch, symmetric and asymmetric CH_2 stretch appeared at 1708, 2853, and 2922 cm^{-1} , respectively. The band at 1283 cm^{-1} slightly shifted the C-O stretch to a low frequency. The characteristic bands of the surfactant molecules slightly shifted to the lower frequency regions, indicating the hydrophobic regions of the monolayer were packed tightly in a crystalline state around the nanoparticle surface.⁷² A distinguished carboxylate (COO) stretch at 1461 cm^{-1} is a confirmation of OA chemisorption on the surface of the nanoparticles *via* its oxygen.^{71,80}

The melting and enthalpy data of lauric acid, TLN, γ -Fe₂O₃, and P- γ -TLN, obtained through DSC, are summarized in Table S3.[†] The sharp endothermic peak at 48.1 °C and the high enthalpy of around 72.4 J g⁻¹ were observed due to the crystallinity of lauric acid. The comparison of thermograms of lauric acid and TLN and the subsequent understanding of the decreased melting point of TLN due to the addition of liquid lipids (Fig. 3B)⁸³ is a unique observation. The nanocarrier P-TLN demonstrates a decrease in enthalpy and melting, which adds to the novelty of our findings.

The results indicated a nanoparticulate transformation, resulting in reduced crystallinity and an imperfect crystal structure compared to primary lipids. The surfactant was found to be a significant factor in this phase transformation, leading to an increased surface-to-volume ratio in smaller particulates. Furthermore, the liquid lipid was observed to distort the crystal structure, weakening crosslinking and thereby reducing the overall melting point.⁸⁴ The disorder in the lipid's crystal structure was calculated using the difference between the melting (T_m) and the onset of melting temperatures (T_o):

$$\text{Disorder of lipid crystallinity} = T_m - T_o \quad (12)$$

The $T_m - T_o$ values serve as a crucial indicator of the disorder in the crystal structure of the lipid matrix. The higher the difference, the more pronounced the disorder. For instance, the solid lipid, lauric acid, exhibited a difference of 2.68, while P- γ -TLN, with a high value of 16.29, depicted a significant level of disturbance. A distinct single melting endotherm emerged at 234.09 °C just before degradation, with an enthalpy of 505.16 J g⁻¹.

The thermogram of $\gamma\text{-Fe}_2\text{O}_3$ underwent an enthalpy change of 1129.7 J g^{-1} and a total weight loss of 15.99% in two stages upon heating the powder sample to 800°C . The first small drop at $\sim 50\text{--}100^\circ\text{C}$ is attributed to the physisorbed water. The weight loss of 9.11% at temperatures ranging from 189.4°C to about 281.7°C is due to multi-layers of water of hydration from the surface of the nanoparticles.⁸⁵ An endothermic peak at 275.9°C corresponds to the transformation of Fe_3O_4 into $\gamma\text{-Fe}_2\text{O}_3$ (Fig. S5†).⁸⁶

The addition of payloads significantly influenced the overall enthalpy and melting behavior of the lipid matrix. Notably, the introduction of PTX at approximately 5 mg resulted in negligible alterations in the lipid matrix's pretransition and main phase transitions, with the exception of a broadening pre-transition with increasing amounts (Fig. 3B).⁸⁷ Conversely, the inclusion of γ -Fe₂O₃ above 10 mg led to a substantial shift and broadening in the phase transition to 52.96 °C and enthalpy of 306.58 J g⁻¹ due to the rigidity, phase transformation, and stabilization of the lipid layers by the encapsulated γ -Fe₂O₃,⁸⁸ necessitating the selection of payloads below this threshold. These findings shed new light on the effects of different payloads on lipid matrix properties.

The peaks were identified using ICSD 00-002-1047 for iron oxide (Fe₂O₃) nanoparticles, providing a clear and reliable method for differentiation. Additional peaks appearing in γ -Fe₂O₃ at 24.25° and 25.52°, which were absent in Fe₃O₄ nanoparticles, served as a clear and distinguishable feature between the two forms (Fig. 3C).⁸⁹ PTX exhibited three sharp peaks at the diffraction angles of 10.02, 12.33, 15.58, 18.63, 21.94, 25.15, and 29.87°, indicating a crystalline structure.⁷³ The XRD patterns of P- γ -TLN clearly revealed the presence of lauric acid with peaks at 20.43, 21.16, 25.17,⁹⁰ and 40.34,⁹¹ while the manifestations attributed to PTX were absent, suggesting its integration within the lipid matrix.⁹² The absence of payload peaks confirmed the amorphous phase transformation, which enhances solubility, surface area, bioavailability, and dissolution rate, providing a clear understanding of the lipid matrix properties.⁹³

3.2.5 In vitro magnetic evaluations. Magnetization measurements depicted that the material analyzed was ferromagnetic with coercivity values of ~200 Ôe and saturation magnetization of 64 emu g⁻¹ (ref. 94) for the γ -Fe₂O₃ nanoparticles. All the samples showed soft ferromagnetic behavior, while the γ -TLN sample shows lower magnetization at higher fields; such reduction might be due to the spin reorientation transition and magnetic anisotropy misalignment.

Fig. 4b illustrates the low-field regions of the $M(H)$ loops. The loops of γ -Fe₂O₃ and γ -TLN were shifted along the field axis,

manifesting an exchange bias (Hex) of 43 Ôe and 36 Ôe for γ -Fe₂O₃ and γ -TLN, respectively. This bias could be attributed to the disordered surface spin as in the already reported chemically synthesized nanoparticles.⁹⁵ Moreover, the disordered surface moments can produce a spin-glass-like state that is effective in pinning the ferromagnetic core⁹⁶ thus producing exchange bias. These findings have significant implications for our understanding of exchange bias in magnetic materials.

Zero field-cooled (ZFC) thermal-demagnetization measurements were carried out at 300–800 K in a 100 Ôe static field (Fig. 4c). The T_c , a crucial parameter, has been depicted as the point of minimum change in dM/dT against the T curve (Fig. 4d). The results show that the T_c of γ -Fe₂O₃ nanoparticles is 644 K, significantly above the therapeutic range required for self-controlled magnetic hyperthermia. This key finding is of utmost importance as it demonstrates the potential of these nanoparticles for therapeutic applications. Encapsulation in a biocompatible agent further improved the T_c , bringing it to the required range without significant magnetization loss.

The obtained results for heating measurements carried out on fluid samples of γ -Fe₂O₃ and γ -TLN in an AC magnetic field of different fields and frequencies are shown in Fig. 4e–h. These graphs illustrated a stable and consistent increase in the temperature over the course of applying a magnetic field for 250 s. This stability in the temperature increase provides reassurance about the reliability of the results. Lipid encapsulated γ -TLN also produced reasonable results at low doses and field exposure time. The rate of increase in temperature was suggestive of an immense SLP value. The SLP represents the magnetothermal behavior of nanoparticles and is estimated by calculating the initial heating rates (dT/dt) ($t = 0$) and applying the equation for the fluid sample:⁹⁷

$$\text{SLP} = C_{\text{water}}(dT/dt)_{(t=0)}V_{\text{sample}}/m(\text{magn.}) \quad (13)$$

C_{water} is the specific heat capacity of the fluid sample (assumed to be equal to that of pure water 4185 J per kg per K), V_{sample} is the sample volume, and $m(\text{magn.})$ is the fraction of the magnetically active elements in the sample. Fig. S6a and b† show that the SLP values of 356 W g⁻¹ were obtained at 518.7 kHz and 23.9 mT, which increased linearly with increasing concentration and strength of the applied magnetic field H_0 . Contrastingly, encapsulated γ -TLN fluid samples manifested a higher magnetothermal response at lower fields and frequencies, i.e., an extreme dependence on applied field amplitude and frequencies.

The AMF produces eddy currents in biological tissues, which can also cause harm to the surrounding healthy tissues measured in terms of the Brezovich criterion. The calculated values for two out of the three frequencies ($13\,051.12\text{ A m}^{-1} \times 163\,800\text{ s}^{-1} = 2.14 \times 10^9\text{ A m}^{-1}\text{ s}^{-1}$, $13\,608.18\text{ A m}^{-1} \times 1\,327\,300\text{ s}^{-1} = 4.45 \times 10^9\text{ A m}^{-1}\text{ s}^{-1}$) used in the study correspond to the criterion i.e., $4.85 \times 10^8\text{ A m}^{-1}\text{ s}^{-1}$, while one falls out ($19\,019.62\text{ A m}^{-1} \times 518\,700\text{ s}^{-1} = 9.87 \times 10^9\text{ A m}^{-1}\text{ s}^{-1}$). Some studies suggest the limit to be less rigid and time-bound,^{98,99} which means that contrary to more than one hour of exposure, 10 minutes pulses would be a safe alternative. However, for the

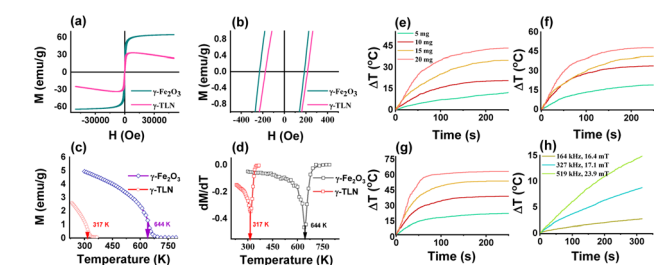


Fig. 4 Room temperature magnetization hysteresis loops of γ -Fe₂O₃ and γ -TLN samples (a), the enlarged low-field portion of the $M(H)$ loops (b), field-cooled thermal demagnetization measurements of γ -Fe₂O₃ nanoparticles in a measuring field of 100 Ôe (c), dM/dT as a function of temperature to calculate the Curie temperature (d). Time-dependent heating curves of γ -Fe₂O₃ fluid samples in a measuring field of (e) $f = 164\text{ kHz}$ and $H_0 = 16.4\text{ mT}$, (f) $f = 327\text{ kHz}$ and $H_0 = 17.09\text{ mT}$, (g) $f = 519\text{ kHz}$ and $H_0 = 23.90\text{ mT}$, and (h) 5 mg γ -Fe₂O₃ encapsulated TLN lyophilized samples measured at different fields and frequencies.



in vivo experimentation, the recommendation for the criterion must be followed to meet the safety standards even for the pulsed treatment.

3.2.6 *In vitro* drug release study. The plain PTX, with its non-ionizable functional groups, exhibits a fascinating characteristic release at the tumor site, which is not significantly influenced by alterations in pH. This pH-independence has important implications for the extent and rate of release, as it suggests a potential advantage in drug delivery systems.¹⁰⁰ The solubility of PTX in the Cremophor EL alcoholic blend is 6 mg mL⁻¹,¹⁰¹ but when this blend is incorporated into an aqueous phase, the solubility decreases with increasing aqueous volume. This intriguing behavior is depicted in Fig. S7A and B,† with the solubility decreasing from 869 to 0.22 µg mL⁻¹ as the concentration and dilution times change from 2.33 to 23.33 and 5 to 50, respectively.

For perfect sink compliance, a concentration and dilution volume at twenty times dilution was selected, with 500 µg mL⁻¹ (1 mL) and 100 mL, respectively. This precision ensures that the solubility of the system remains well below the saturation solubility (6.49%) of the twenty times dilution, *i.e.*, 77.0 µg mL⁻¹, maintaining perfect sink conditions. The temperature-dependent PTX release was observed at different simulated temperatures, including 37 °C (body temperature), 41 °C (tumor microenvironment), and 45 °C (magnetic hyperthermia). These results were compared with that of the PTX solution in the Cremophor EL ethanol blend, which showed a maximum release of 99.8% in 72 h (Fig. 5A).

The nanoparticle demonstrated a temperature-dependent release pattern, with the release order being 45 °C > 41 °C > 37 °C. The release at 45 °C and 41 °C were comparable due to the solid-liquid phase transition beyond ~39 °C. This release pattern at 41 °C and 45 °C, indicating that the reopening of pores caused PTX release with a temperature-dependent diffusion process, confirms the suitability of P-γ-TLN for thermos-responsive drug delivery under hyperthermia. The release rates at the later stages of the drug release profile illustrated a slower rate attributed to the decrease in the concentration gradient, providing further evidence of the system's potential.¹⁰²

The release profiles of PTX with/without AMF for 12.16 h are depicted in Fig. 5B. The extent of drug release without a magnetic field pulse was relatively lower, *i.e.*, 17.28% ± 2.38% at pH 7.4. The core contains most of the lipophilic payload, but

the outer part has some of the liquid lipid trapping payload in the periphery, thus releasing a small portion due to erosion. Proceeding this release, the slow, sustained release continued for the rest of the time course.¹⁰³

In comparison, under a stronger magnetic field, a 2.553-fold increase in PTX release (44.13% ± 0.56%) was observed at pH 7.4 (Fig. 5B). The magnetic-field specified drug release profile refers to the drug release pattern that is specifically influenced by the application of a magnetic field. This profile suggested an effective burst due to system collapse from the lipid matrix of P-γ-TLN.¹⁰⁴ The pulse during the ON cycle was given for a moment of 10 min after which the system recovered back to the micro-emulsion state and subsequently reduced the diffusivity from the collapsed matrix, resulting in slower release rates during the OFF cycles. Also, during this phase, the system temperature decreases, resulting in solidifying the matrix to halt the drug leakage.¹⁰⁵

3.2.7 Stability studies. The particle size of the P-γ-TLN 12 stored at 5 ± 3 °C ranged between 183.1 and 193.1 nm, and at 25 ± 2 °C, the depicted size ranges were between 183.1 nm and 191.6 nm, respectively (Table S4†). The particle size increased during storage, maintained at a change percentage of approximately 5%. Particle size increases at elevated temperatures can be attributed to the raised kinetic energy of the colloidal system, resulting in continuous collusion and, subsequently, aggregation.¹⁰⁶ In addition, the low and relatively constant polydispersity changes of 3.79% and 4.33% were observed for all formulations stored at 5 ± 3 °C for 12 months and 25 ± 2 °C for 6 months, respectively. The entrapment efficiency depicted that the PTX experienced approximately a 5% decrease, whereas γ-Fe₂O₃ depicted 7.69% and 11.82% for long-term and accelerated stability conditions. This might be due to the interaction of magnetic moments with electric and metallic noise factors.

3.2.8 Cell lines and cell culture

3.2.8.1 *In vitro* cytotoxicity. The cytotoxicity findings of PTX, γ-Fe₂O₃, TLN, P-TLN, and P-γ-TLN were investigated by the MTT test using the MCF-7 cell lines, presented in Fig. 6. Even at the highest concentration of 10 mg mL⁻¹, 43.32% ± 1.11%

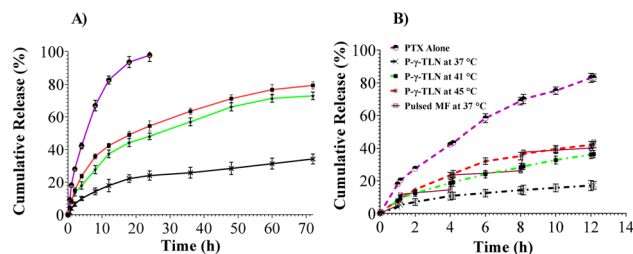


Fig. 5 *In vitro* PTX release assessment using temperature as a parameter for plain PTX and PTX-loaded TLN (A). *In vitro* PTX release assessment over 12.16 h under an alternating magnetic field (solid lines) and without a magnetic field (dotted lines) (B).

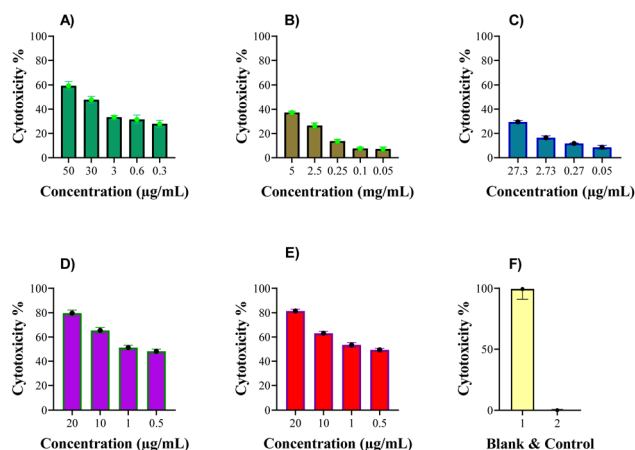


Fig. 6 Cell viability measurements at concentrations predetermined in the established protocols for PTX (A), γ-Fe₂O₃ (B), TLN (C), P-TLN (D), P-γ-TLN (E), blank, and control (F).



cytotoxicity was detected with the calculated IC₅₀ of 11.41 mg mL⁻¹ for γ -Fe₂O₃ (Fig. 6B). Furthermore, more than 90% of the cells were still alive at 0.1 mg mL⁻¹ of γ -Fe₂O₃ concentration, demonstrating that they were cytocompatible. Similarly, TLN depicted cytocompatibility due to 29.43 ± 1.171 cell cytotoxicity at 27.33 μ g mL⁻¹ concentration with an IC₅₀ of 54.98 μ g mL⁻¹ (Fig. 6D). These results agreed with the previous finding on stearic/OA nanostructured lipid-based systems with the IC₅₀ value of 44.6 μ g mL⁻¹.¹⁰⁷ When cells were treated with P-TLN and P- γ -TLN, the vitality of the cells decreased dramatically, suggesting a higher level of inhibitory activity of cancer cells compared to free PTX and γ -Fe₂O₃ alone (Fig. 6E and F).¹⁰⁸

The IC₅₀ values of P-TLN and P- γ -TLN were approximately 34-fold lower than the free PTX, accounting for the drug's enhanced solubility and cytotoxic activity in TLN compared with the aqueous solution. Positive and standard control depicted 99.41 ± 0.092 and 0.00035% ± 0.34% cell cytotoxicity, respectively (Fig. 6F).

3.2.8.2 In vitro chemo-magnetic hyperthermic cytotoxicity. PTX and γ -Fe₂O₃ were used as a control to determine the cytocompatibility with and without AMF, respectively. PTX produced similar cytotoxicity results with/without AMF (Fig. 6A and 7A), whereas γ -Fe₂O₃/AMF produced concentration-dependent cytotoxicity due to the enhanced magnetic hyperthermia and cell aggregation/agglomerations at higher levels.¹⁰⁹

The concentration of γ -Fe₂O₃ at around 125 μ g mL⁻¹ produced a minimal cytotoxicity of about 7%, which was consistent with the results depicted in Fig. 4, *i.e.*, concentration-dependent ΔT at all field strengths (Fig. 7C).

At a higher concentration of 1000 μ g mL⁻¹, ~80% cytotoxicity was achieved, which could also be due to enhanced magnetic heating and loss in colloidal stability of the dispersion.¹¹⁰ P- γ -TLN had similar results in AMF to that of P-TLN due to the equal amount of PTX and γ -Fe₂O₃ loaded in them, *i.e.*, 0.5–20 μ g mL⁻¹ (Fig. 7A and D). However, when the concentration of γ -Fe₂O₃ was increased to 500 μ g mL⁻¹, a significant

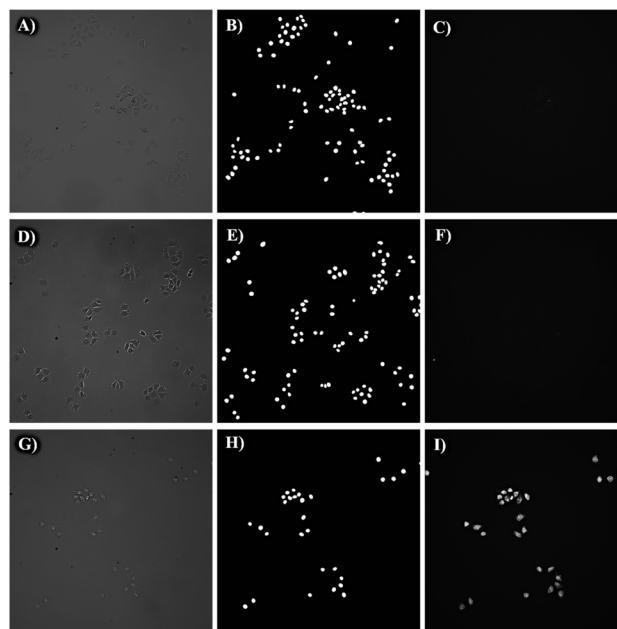


Fig. 8 Cell uptake analyses for normal groups (A)–(C), rhodamine only (D)–(F), and rhodamine-loaded nanoparticles (G)–(I). Bright field, blue, and green fluorescent imaging from left to right, depicting internalization inside the cells.

enhancement in cytotoxicity was observed at all concentrations of the PTX in P- γ -TLN, a result of significant impact in our research, indicating almost 98% cell cytotoxicity at 20 μ g mL⁻¹ of PTX.

3.2.8.3 In vitro cell uptake imaging. Microscopy studies, as shown in Fig. 8, show that the cells could uptake green-fluorescent rhodamine-B in the form of nanoparticles compared to the solution. Higher fluorescence intensity, corresponding to 50 μ M RHO concentration, distributed throughout the cytoplasm of the cells. Similarly, blue fluorescent-DAPI accumulated in the nucleus of the MCF-7 cells. Strong co-localization was observed with rhodamine-loaded nanoparticles due to a stable environment that protects the encapsulated fluorescent molecules from degradation by enzymes or pH changes in the cellular environment.¹¹¹ In solution, the same molecules might degrade or lose activity before being internalized. Also, cellular uptake was enhanced through endocytosis mechanisms for which the lipid-based nature mimics cell membranes, facilitating better interaction and uptake than free molecules in solutions. Cellular efflux mechanisms might more readily expel free molecules in solution, whereas encapsulation in NLCs can protect them from immediate recognition and expulsion.

4. Conclusions

In this work, a new strategy involving chemo-magnetic thermoresponsive delivery, based on the co-loading of a chemotherapeutic agent, PTX, and γ -Fe₂O₃ nanoparticles, has been accomplished. We have successfully optimized and demonstrated the biocompatible thermoresponsive performance of

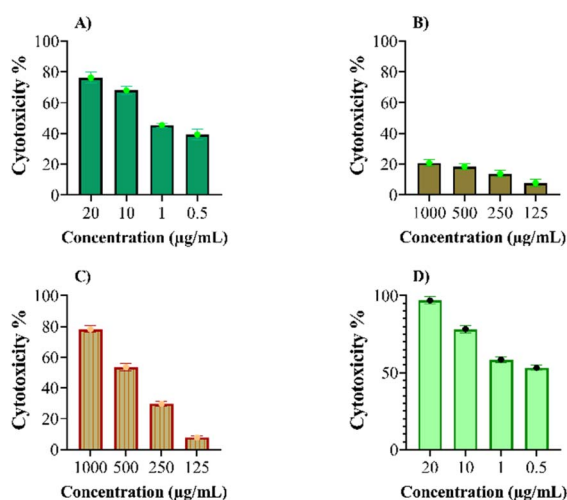


Fig. 7 Cell toxicity measurements under AMF at concentrations pre-determined in the established protocols for PTX (A), γ -Fe₂O₃ (B), γ -Fe₂O₃/AMF (C), and P- γ -TLN/AMF (D).



TLN under clinically safe conditions of magnetically clickable hyperthermia. The tissue hyperthermia-triggered release was justified with an increased PTX release at elevated temperatures. Similarly, an enhanced magnetically triggered release (~2.5 fold) was also recorded using an AMF device. The successful validation of the release behavior through cytotoxicity assessment with/without AMF further reassures the efficacy of this system. The AMF-triggered cytotoxicity testing proved this system to be efficacious in killing almost all the cells (~98%), but further investigations are required on suitable animal models in the future.

Data availability

The data supporting this article have been included as part of the ESI.†

Author contributions

Muhammad Tayyab: conceptualization, methodology, visualization, investigation, formal analysis, writing – original draft, writing – review & editing. Naveed Ahmed: conceptualization, supervision, visualization, formal analysis, software, review & editing. M. Hisham Al Nasir: methodology, investigation, formal analysis, writing – review & editing. Ayesha Ihsaan: methodology, visualization, investigation, writing – review & editing. Asim Ur Rehman: methodology, visualization, investigation, formal analysis, writing – review & editing, resources. Khurram Shahzad Munawar: methodology, investigation, validation, review & editing, resources. Mubashar Rehman: supervision, conceptualization, methodology, visualization, validation, funding acquisition, writing – original draft, writing – review & editing, resources, project administration. Maria del Carmen Iglesias de la Cruz: methodology, visualization, validation, writing – review & editing, resources. Daniel Jaque Garcia: supervision, conceptualization, methodology, visualization, validation, funding acquisition, writing – review & editing, resources, project administration.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 T. Li, H. Zhang, M. Lian, Q. He, M. Lv, L. Zhai, J. Zhou, K. Wu and M. Yi, *J. Hematol. Oncol.*, 2025, **18**(1), 1–25.
- 2 American Cancer Society, *Breast Cancer Facts & Figures*, American Cancer Society, 2024.
- 3 S. Guo, M. Vieweger, K. Zhang, H. Yin, H. Wang, X. Li, S. Li, S. Hu, A. Sparreboom and B. M. Evers, *Nat. Commun.*, 2020, **11**, 972.
- 4 J. Marin, R. Al-Abdulla, E. Lozano, O. Briz, L. Bujanda, J. M. Banales and R. Macias, *Anti-Cancer Agents Med. Chem.*, 2016, **16**, 318–334.
- 5 L. Zhang, B.-Z. Zhai, Y.-J. Wu and Y. Wang, *Drug Delivery*, 2023, **30**, 1–18.
- 6 E. G. Fuller, H. Sun, R. D. Dhavalikar, M. Unni, G. M. Scheutz, B. S. Sumerlin and C. Rinaldi, *ACS Appl. Polym. Mater.*, 2019, **1**(2), 211–220.
- 7 A. Shakeri-Zadeh and J. W. M. Bulte, *Nat. Rev. Bioeng.*, 2025, **3**, 245–260.
- 8 P. C. Lyon, M. D. Gray, C. Mannaris, L. K. Folkes, M. Stratford, L. Campo, D. Y. Chung, S. Scott, M. Anderson and R. Goldin, *Lancet Oncol.*, 2018, **19**, 1027–1039.
- 9 N. Borys and M. W. Dewhurst, *Adv. Drug Delivery Rev.*, 2021, **178**, 113985.
- 10 M. Regenold, K. Kaneko, X. Wang, H. B. Peng, J. C. Evans, P. Bannigan and C. Allen, *J. Controlled Release*, 2023, **354**, 19–33.
- 11 M. Bhuimali, S. Munshi, K. Hapa, P. K. Kadu and P. P. Kale, *Int. J. Polym. Mater. Polym. Biomater.*, 2024, **73**, 395–404.
- 12 Y. Dou, K. Hynynen and C. Allen, *J. Controlled Release*, 2017, **249**, 63–73.
- 13 N. Borys and M. W. Dewhurst, *Adv. Drug Delivery Rev.*, 2021, **178**, 113985.
- 14 M. Regenold, K. Kaneko, X. Wang, H. B. Peng, J. C. Evans, P. Bannigan and C. Allen, *J. Controlled Release*, 2023, **354**, 19–33.
- 15 M. Rehman, A. Ihsan, A. Madni, S. Z. Bajwa, D. Shi, T. J. Webster and W. S. Khan, *Int. J. Nanomed.*, 2017, **8325–8336**.
- 16 M. Anwar, M. Rehman, T. Ur-Rehman, M. I. Khan, N. Ahmed, A. Madni and M. Tayyab, *J. Pharm. Sci.*, 2025, **103723**.
- 17 M. Yoozbashi, H. Rashidzadeh, M. Kermanian, S. Sadighian, M.-J. Hosseini, Z. Kaboli and K. Rostamizadeh, *J. Biomater. Appl.*, 2022, **36**, 1055–1063.
- 18 G. Qing, X. Zhao, N. Gong, J. Chen, X. Li, Y. Gan and X. J. Liang, *Nat. Commun.*, 2019, **10**(1), 4336.
- 19 I. Brezaniova, M. Hruby, J. Kralova, V. Kral, Z. Cernochova, P. Cernoch and P. Stepanek, *J. Controlled Release*, 2016, **241**, 34–44.
- 20 M. Kawish, S. Ullah, T. Roome, A. Razzak, S. Aslam and M. Raza Shah, *Pharm. Dev. Technol.*, 2024, **29**(7), 762–775.
- 21 Y. N. Tan, Y. P. Li, J. D. Huang, M. Luo, S. S. Li, A. W. M. Lee and X. Y. Guan, *Cancer Lett.*, 2021, **522**, 238–254.
- 22 R. Massart, *IEEE Trans. Magn.*, 1981, **17**, 1247–1248.



- 23 S. Zhu, Y. Leng, M. Yan, X. Tuo, J. Yang, L. Almásy, Q. Tian, G. Sun, L. Zou and Q. Li, *Appl. Surf. Sci.*, 2018, **447**, 381–387.
- 24 M. E. Villanueva, F. Giudice, E. Ambroggio and R. V. Vico, *Langmuir*, 2021, **37**, 1861–1873.
- 25 P. Banerjee, T. Geng, A. Mahanty, T. Li, L. Zong and B. Wang, *Int. J. Pharm.*, 2019, **557**, 374–389.
- 26 C. N. Njoku and S. K. Otisi, in *Response Surface Methodology-Research Advances and Applications*, IntechOpen, 2023.
- 27 S. N. A. Bukhari, A. Ali, M. A. Hussain, M. Tayyab, N. F. Alotaibi, M. A. Elsherif, K. Junaid and H. Ejaz, *Polymers*, 2022, **14**, 1904.
- 28 G. A. El-Emam, A. Motawea, W. E. Abd El Hady, S. Saber, A. A. Mourad, H. A. Ramadan and A. M. El-Baz, *J. Drug Delivery Sci. Technol.*, 2023, **89**, 105012.
- 29 S. A. Lee, H. J. Joung, H. J. Park and G. H. Shin, *J. Food Sci.*, 2017, **82**, 904–912.
- 30 N. Kumar, P. K. S. Rathore, R. Sharma and N. K. Gupta, *Appl. Therm. Eng.*, 2023, **224**, 120088.
- 31 R. M. Iacobazzi, F. Vischio, I. Arduino, F. Canepa, V. Laquintana, M. Notarnicola, M. P. Scavo, G. Bianco, E. Fanizza and A. A. Lopodota, *J. Colloid Interface Sci.*, 2022, **608**, 239–254.
- 32 S. Ghosh, I. Ghosh, M. Chakrabarti and A. Mukherjee, *Food Chem. Toxicol.*, 2020, **136**, 110989.
- 33 R. Nasiri, J. H. Almaki, A. B. Idris, F. A. A. Majid, M. Nasiri, M. Salouti, M. Irfan, N. Amini and M. Marvibaigi, *Mater. Sci. Eng., C*, 2016, **69**, 1147–1158.
- 34 F. Marino, E. R. Lopez, A. Arnosa, M. A. G. Gómez, Y. Pineiro, J. Rivas, C. Alvarez-Lorenzo and J. Fernandez, *J. Mol. Liq.*, 2022, **348**, 118401.
- 35 N. Kumar, S. Goindi, B. Saini and G. Bansal, *J. Therm. Anal. Calorim.*, 2014, **115**, 2375–2383.
- 36 A. Ahmed, Y. Chaker, E. H. Belarbi, O. Abbas, J. Chotard, H. Abassi, A. N. Van Nhien, M. El Hadri and S. Bresson, *J. Mol. Struct.*, 2018, **1173**, 653–664.
- 37 S. Khizar, N. M. Ahmad, N. Ahmed, S. Manzoor and A. Elaissari, *Polym. Adv. Technol.*, 2020, **31**, 1732–1743.
- 38 G. Kandasamy, A. Sudame, T. Luthra, K. Saini and D. Maity, *ACS Omega*, 2018, **3**, 3991–4005.
- 39 S. Lv, Z. Tang, D. Zhang, W. Song, M. Li, J. Lin, H. Liu and X. Chen, *J. Controlled Release*, 2014, **194**, 220–227.
- 40 H. Oliveira, E. Pérez-Andrés, J. Thevenot, O. Sandre, E. Berra and S. Lecommandoux, *J. Controlled Release*, 2013, **169**, 165–170.
- 41 M. Ferreira, L. L. Chaves, S. A. C. Lima and S. Reis, *Int. J. Pharm.*, 2015, **492**, 65–72.
- 42 M. Jamil, I. S. Mustafa, N. M. Ahmed and S. B. S. Hamid, *Biomater. Adv.*, 2022, **143**, 213178.
- 43 F. Ganjoudadeh, S. Khorrami and S. Gharbi, *J. Drug Delivery Sci. Technol.*, 2022, **71**, 103340.
- 44 W. Montha, W. Maneepprakorn, I.-M. Tang and W. Pon-On, *RSC Adv.*, 2020, **10**, 40206–40214.
- 45 P. Parhi and S. K. Sahoo, *J. Colloid Interface Sci.*, 2015, **451**, 198–211.
- 46 L. M. Negi, M. Jaggi and S. Talegaonkar, *Int. J. Pharm.*, 2014, **461**, 403–410.
- 47 S. N. A. S. Azhar, S. E. Ashari, J. K. Tan, N. K. Kassim, M. Hassan, N. Zainuddin, R. Mohamad and I. D. M. Azmi, *Ind. Crops Prod.*, 2023, **198**, 116668.
- 48 Y.-k. Sun, M. Ma, Y. Zhang and N. Gu, *Colloids Surf., A*, 2004, **245**, 15–19.
- 49 Y. M. Gabal, A. O. Kamel, O. A. Sammour and A. H. Elshafeey, *Int. J. Pharm.*, 2014, **473**, 442–457.
- 50 T. V. Truong, M. Ghosh, B. Hosamani, T. V. Baiju, G. Dhandapani, E. Wachtel, E. Kesselman, D. Danino, M. Sheves and I. N. Namboothiri, *J. Pept. Sci.*, 2019, **25**, e3174.
- 51 M. A. Hussain, A. Ali, T. G. Alsahli, N. Khan, A. Sharif, M. T. Haseeb, O. A. Alsaidan, M. Tayyab and S. N. A. Bukhari, *Gels*, 2023, **9**, 525.
- 52 Z. Azizi, R. Barzegarian and M. Behvandi, *Sustainable Energy Technologies and Assessments*, 2022, **50**, 101876.
- 53 U. K. Baruah, K. Gowthamarajan, V. Ravisankar, V. V. S. R. Karri, P. K. Simhadri and V. Singh, *J. Drug Targeting*, 2018, **26**, 576–591.
- 54 L. A. S. Bahari and H. Hamishehkar, *Adv. Pharm. Bull.*, 2016, **6**, 143.
- 55 M. T. Chaichan, M. T. Mahdi, H. A. Kazem, A. H. Al-Waeli, M. A. Fayad, A. A. Al-Amiery, W. N. R. W. Isahak, A. A. H. Kadhum and M. S. Takriff, *Sustainability*, 2022, **14**, 12015.
- 56 M. I. Teixeira, C. M. Lopes, H. Gonçalves, J. Catita, A. M. Silva, F. Rodrigues, M. H. Amaral and P. C. Costa, *J. Mol. Liq.*, 2023, **388**, 122749.
- 57 R. Bnyan, I. Khan, T. Ehtezazi, I. Saleem, S. Gordon, F. O'Neill and M. Roberts, *J. Pharm. Sci.*, 2018, **107**, 1237–1246.
- 58 A. Alhalmi, S. Amin, S. Beg, R. Al-Salahi, S. R. Mir and K. Kohli, *J. Drug Delivery Sci. Technol.*, 2022, **74**, 103590.
- 59 N. Dudhipala and K. Veerabrahma, *Eur. J. Pharm. Biopharm.*, 2017, **110**, 47–57.
- 60 L. K. Holte, B. A. Kuran, G. L. Richmond and K. E. Johnson, *J. Phys. Chem. C*, 2014, **118**, 10024–10032.
- 61 U. Klinkesorn and Y. Namatsila, *Food Hydrocolloids*, 2009, **23**, 1374–1380.
- 62 P. Witayaudom and U. Klinkesorn, *J. Colloid Interface Sci.*, 2017, **505**, 1082–1092.
- 63 X. Song, Y. Zhao, S. Hou, F. Xu, R. Zhao, J. He, Z. Cai, Y. Li and Q. Chen, *Eur. J. Pharm. Biopharm.*, 2008, **69**, 445–453.
- 64 P. Kaur, T. Garg, G. Rath, R. R. Murthy and A. K. Goyal, *Drug Delivery*, 2016, **23**, 1912–1925.
- 65 S. Bhattacharjee, *J. Controlled Release*, 2016, **235**, 337–351.
- 66 A. D. Wulandari, S. Sutriyo and R. Rahmasari, *J. Adv. Pharm. Technol. Res.*, 2022, **13**(2), 89–94.
- 67 S. Jose, S. S. Anju, T. A. Cinu, N. A. Aleykutty, S. Thomas and E. B. Souto, *Int. J. Pharm.*, 2014, **474**(1–2), 6–13.
- 68 A. A. Shimojo, A. R. V. Fernandes, N. R. Ferreira, E. Sanchez-Lopez, M. H. Santana and E. B. Souto, *Antioxidants*, 2019, **8**(8), 272.
- 69 M. Khaledian, M. S. Nourbakhsh, R. Saber, H. Hashemzadeh and M. H. Darvishi, *Int. J. Nanomed.*, 2020, 6167–6182.



- 70 A. Carvalho, M. Martins, M. Corvo and G. Feio, *Mater. Sci. Eng., C*, 2014, **43**, 521–526.
- 71 X. Fu, W. Kong, Y. Zhang, L. Jiang, J. Wang and J. Lei, *RSC Adv.*, 2015, **5**, 68881–68889.
- 72 M. I. Idris, J. Zaloga, R. Detsch, J. A. Roether, H. Unterweger, C. Alexiou and A. R. Boccaccini, *Sci. Rep.*, 2018, **8**, 7286.
- 73 L. Zhang, R. He and H.-C. Gu, *Appl. Surf. Sci.*, 2006, **253**, 2611–2617.
- 74 S. Alipour, H. Montaseri and M. Tafaghodi, *Colloids Surf., B*, 2010, **81**, 521–529.
- 75 K. Hasanbegloo, S. Banihashem, B. F. Dizaji, S. Bybordi, N. Farrokh-Eslamlou, P. G.-s. Abadi, F. S. Jazi and M. Irani, *Int. J. Biol. Macromol.*, 2023, **230**, 123380.
- 76 L. Wang, X. Zhao, F. Yang, W. Wu, M. Wu, Y. Li and X. Zhang, *Int. J. Biol. Macromol.*, 2019, **138**, 207–214.
- 77 M. M. Bora, P. Gogoi, D. C. Deka and D. K. Kakati, *Ind. Crops Prod.*, 2014, **52**, 721–728.
- 78 D. B. Alfaro, E. Prokhorov, G. L. Barcenar and H. García, *J. Mol. Liq.*, 2023, **374**, 121304.
- 79 E. Darezereshki, *Mater. Lett.*, 2010, **64**, 1471–1472.
- 80 W. K. Lewis, A. T. Rosenberger, J. R. Gord, C. A. Crouse, B. A. Harruff, K. S. Fernando, M. J. Smith, D. K. Phelps, J. E. Spowart and E. A. Gulians, *J. Phys. Chem. C*, 2010, **114**, 6377–6380.
- 81 I. D. R. Pedro, O. P. Almeida, H. R. Martins, J. de Alcântara Lemos, A. L. B. de Barros, E. A. Leite and G. Carneiro, *J. Drug Delivery Sci. Technol.*, 2019, **54**, 101370.
- 82 V. H. S. Araujo, P. B. da Silva, I. O. Szlachetka, S. W. da Silva, B. Fonseca-Santos, M. Chorilli, R. Ganassin, G. R. T. de Oliveira, M. C. O. da Rocha and R. P. Fernandes, *Colloids Surf., A*, 2020, **602**, 125070.
- 83 C. Dhanya, W. Paul, M. Rekha and R. Joseph, *J. Mol. Liq.*, 2023, **380**, 121738.
- 84 R. Osanlou, M. Emtiazjoo, A. Banaei, M. A. Hesarinejad and F. Ashrafi, *Colloids Surf., A*, 2022, **641**, 128588.
- 85 S. J. Iyengar, M. Joy, C. K. Ghosh, S. Dey, R. K. Kotnala and S. Ghosh, *RSC Adv.*, 2014, **4**, 64919–64929.
- 86 M. Aliahmad and N. Nasiri Moghaddam, *Mater. Sci.-Pol.*, 2013, **31**, 264–268.
- 87 M. Ali, P. Adie, C. H. Marrows, D. Greig, B. J. Hickey and R. L. Stamps, *Nat. Mater.*, 2007, **6**, 70–75.
- 88 C. C. Cheung, I. Monaco, N. Kostevšek, M. C. Franchini and W. T. Al-Jamal, *Colloids Surf., B*, 2021, **198**, 111453.
- 89 W. Kim, C.-Y. Suh, S.-W. Cho, K.-M. Roh, H. Kwon, K. Song and I.-J. Shon, *Talanta*, 2012, **94**, 348–352.
- 90 W. Kong, Z. Liu, Y. Yang, C. Zhou and J. Lei, *Constr. Build. Mater.*, 2017, **152**, 568–575.
- 91 J. Liu, D. Jiang, H. Fei, Y. Xu, Z. Zeng and W. Ye, *ACS Omega*, 2021, **6**, 23542–23550.
- 92 B. Rodenak-Kladniew, N. Noacco, I. P. de Berti, S. J. Stewart, A. F. Cabrera, V. A. Alvarez, M. G. de Bravo, N. Durán, G. R. Castro and G. A. Islan, *Colloids Surf., B*, 2021, **202**, 111710.
- 93 N. M. Patro, K. Devi, R. S. Pai and S. Suresh, *J. Nanopart. Res.*, 2013, **15**, 1–18.
- 94 D. Cao, H. Li, L. Pan, J. Li, X. Wang, P. Jing, X. Cheng, W. Wang, J. Wang and Q. Liu, *Sci. Rep.*, 2016, **6**, 32360.
- 95 H. Khurshid, W. Li, M.-H. Phan, P. Mukherjee, G. C. Hadjipanayis and H. Srikanth, *Appl. Phys. Lett.*, 2012, **101**(2), 022403.
- 96 H. Wang, T. Zhu, K. Zhao, W. Wang, C. Wang, Y. Wang and W. Zhan, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2004, **70**, 092409.
- 97 A. Hanini, L. Lartigue, J. Gavard, K. Kacem, C. Wilhelm, F. Gazeau, F. Chau and S. Ammar, *J. Magn. Magn. Mater.*, 2016, **416**, 315–320.
- 98 M. T. Luiz, J. A. P. Dutra, J. S. R. Viegas, J. T. C. de Araújo, A. G. Tavares Junior and M. Chorilli, *Pharmaceutics*, 2023, **15**(3), 751.
- 99 M. K. Kwok, C. C. Maley, A. Dworkin, S. Hattersley, P. Southern and Q. A. Pankhurst, *Appl. Phys. Lett.*, 2023, **122**(24), 240502.
- 100 J. M. Terwogt, B. Nuijen, W. T. B. Huinink and J. Beijnen, *Cancer Treat. Rev.*, 1997, **23**, 87–95.
- 101 B. Nuijen, M. Bouma, J. H. Schellens and J. H. Beijnen, *Invest. New Drugs*, 2001, **19**, 143–153.
- 102 G. Tan, S. Yu, J. Li and W. Pan, *Int. J. Biol. Macromol.*, 2017, **103**, 941–947.
- 103 G.-H. Son, B.-J. Lee and C.-W. Cho, *J. Pharm. Invest.*, 2017, **47**, 287–296.
- 104 N. S. Satarkar and J. Z. Hilt, *J. Controlled Release*, 2008, **130**, 246–251.
- 105 W.-C. Huang, S.-H. Hu, K.-H. Liu, S.-Y. Chen and D.-M. Liu, *J. Controlled Release*, 2009, **139**, 221–228.
- 106 F.-Q. Hu, S.-P. Jiang, Y.-Z. Du, H. Yuan, Y.-Q. Ye and S. Zeng, *Int. J. Pharm.*, 2006, **314**, 83–89.
- 107 S. Abdollahpour, N. Mahdieh, Z. Jamali, A. Akbarzadeh, T. Toliyat and M. Paknejad, *BioNanoScience*, 2017, **7**, 32–39.
- 108 R. Al-Obaidy, A. J. Haider, S. Al-Musawi and N. Arsad, *Sci. Rep.*, 2023, **13**, 3180.
- 109 A. E. Deatsch and B. A. Evans, *J. Magn. Magn. Mater.*, 2014, **354**, 163–172.
- 110 H. L. Rodríguez-Luccioni, M. Latorre-Esteves, J. Méndez-Vega, O. Soto, A. R. Rodríguez, C. Rinaldi and M. Torres-Lugo, *Int. J. Nanomed.*, 2011, 373–380.
- 111 X. Shao, C. Meng, W. Song, T. Zhang and Q. Chen, *Adv. Drug Delivery Rev.*, 2023, 114977.

