

REVIEW

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New paradigms of 2D layered material self-driven photodetectors

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By virtue of the high carrier mobility, diverse electronic band structures, excellent electrostatic tunability, easy integration, and strong light-harvesting capability, 2D layered materials (2DLMs) have emerged as compelling contenders in the realm of photodetection and ushered in a new era of optoelectronic industry. In contrast to powered devices, self-driven photodetectors boast a wealth of advantages, notably low dark current, superior signal-to-noise ratio, low energy consumption, and exceptional compactness. Nevertheless, the construction of self-driven 2DLM photodetectors based on traditional p–n, homo-type, or Schottky heterojunctions, predominantly adopting a vertical configuration, confronts insurmountable dilemmas such as intricate fabrication procedures, sophisticated equipment, and formidable interface issues. In recent years, worldwide researchers have been devoted to pursuing exceptional strategies aimed at achieving the self-driven characteristics. This comprehensive review offers a methodical survey of the emergent paradigms toward self-driven photodetectors constructed from 2DLMs. Firstly, the burgeoning approaches employed to realize diverse self-driven 2DLM photodetectors are compiled, encompassing strategies such as strain modulation, thickness tailoring, structural engineering, asymmetric ferroelectric gating, asymmetric contacts (including work function, contact length, and contact area), ferroelectricity-enabled bulk photovoltaic effect, asymmetric optical antennas, among others, with a keen eye on the fundamental physical mechanisms that underpin them. Subsequently, the prevalent challenges within this research landscape are outlined, and the corresponding potential approaches for overcoming these obstacles are proposed. On the whole, this review highlights new device engineering avenues for the implementation of bias-free, high-performance, and highly integrated 2DLM optoelectronic devices.

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1. Introduction

The revolutionary exfoliation of graphene by Novoselov & Geim in 2004 opened the prelude to the exploration of two-dimensional layered materials (2DLMs).¹ Broadly speaking, 2DLMs encompass ultrathin nanosheets/nanofilms composed of a solitary monolayer or a few monolayer motifs held together by feeble van der Waals interactions.^{2–14} Up to now, 2DLMs have exhibited a host of unique physical and chemical properties, such as high surface-to-volume ratio, large interlayer spacing, easy integration, excellent flexibility, *etc.*, which make them

widely applicable in extensive research fields spanning electronics,^{15–17} information storage,¹⁸ photonics,^{19–25} medical therapy,²⁶ energy storage,^{27–30} catalysis,^{31–33} electromagnetic shielding,³⁴ assisted driving,³⁵ superconductivity,^{36–38} *etc.*

In recent years, by virtue of the high carrier mobility, diverse electronic band structures, excellent electrostatic tunability, and strong light-harvesting capability, 2DLMs have been highly regarded and demonstrated remarkable potential in the field of photodetection.^{39–46} Generally speaking, 2DLM photodetectors can be largely divided into two main categories, including powered devices and self-driven devices. Powered photodetectors are devices that necessitate the impetus of external auxiliary power sources for normal operation.^{47–66} The most common powered photodetectors include metal–semiconductor (semimetal)–metal (MSM) photoconductors and phototransistors.^{67,68} By contrast, self-driven photodetectors refer to devices that can operate normally without the propulsion of any external power sources. Compared to powered devices, self-driven photodetectors bear a series of advantages. To begin with, due to the absence of an external driving electric field, the dark current of the self-

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driven devices is extremely low, resulting in a remarkable signal-to-noise ratio and low energy consumption. Secondly, devoid of the intricate power systems, self-driven photodetectors typically exhibit simpler constructions, rendering them not only more compact and reliable but also significantly easier to maintain. This proves exceptionally advantageous in circumstances where a sustained power supply to the photodetectors is impractical, notably in realms such as remote sensing, wearable electronic devices, and Internet of Things (IoT). Thirdly, owing to the exemption of any power supply, the electromagnetic radiation emitted during the operation of self-driven photodetectors is notably feeble, making them more difficult to be detected. This exceptional level of concealment is paramount across diverse fields such as military reconnaissance and intelligence gathering, helping to reduce the risk of exposure and bolster the mission success rate. Furthermore, apart from the incident light energy, self-driven photodetectors operate without consuming electricity or any other forms of energy, which is highly environmentally friendly.

Basically, a bias-free photodetector is mainly constructed based on a p-n junction^{69–72} or a Schottky junction.^{73–75} However, self-driven photodetectors built of traditional p-n or Schottky junctions, which are mostly in a vertical form, suffer from a host of shortcomings. To begin with, due to the extremely thin thickness, 2DLMs are far less robust as compared to bulk counterparts. In this consideration, traditional doping processes, such as ion implantation, are quite harmful to 2DLMs. To date, the mature technologies of chemical doping for 2DLMs are not yet established. Secondly, it is difficult to precisely control the doping depth by exploiting traditional doping techniques as for the atomically thin materials, making it challenging to achieve controllable preparation of atomically thin p-n junctions. Thirdly, the output signal of p-n junctions is intrinsically limited by the Shockley–Queisser limit. Therefore, it is impending to relentlessly resort to novel schemes for overcoming the above difficulties.

In recent years, researchers all around the world have been striving to exploring alternative strategies for realizing self-

driven photodetection based on 2DLMs through device engineering. For example, in 2018, Zhou *et al.* demonstrated zero-bias metal–semiconductor–metal WSe₂ photodetectors by exploiting asymmetric contact geometries.⁷⁶ In 2022, Lu *et al.* developed self-driven MoS₂ homojunction photodetector by utilizing localized strain engineering.⁷⁷ Most recently, by exploiting thickness tailoring, Hu *et al.* have successfully fabricated a bias-free MoT₂ homojunction photodetector array.⁷⁸

Despite that encouraging research progress has been achieved, the preceding research findings are extensive but relatively scattered. In addition, the underlying physical principles for implementing the self-driven operation of different studies are notably distinct. These diverse ingredients pose tricky dilemmas for researchers in swiftly attaining a comprehensive and profound understanding on this rapidly expanding research domain.

In this review, a systematic overview on the new device engineering paradigms of self-driven photodetectors built of 2DLMs has been provided. Firstly, the emerging strategies for realizing various self-driven 2DLM photodetectors have been epitomized, including strain modulation, thickness tailoring, structural engineering, asymmetric ferroelectric gating, asymmetric contacts (including work function, contact length, and contact area), ferroelectricity-enabled bulk photovoltaic effect, asymmetric optical antennas, *etc.*, with the elaboration on the fundamental physical principles. Following this, the prevailing challenges standing in the way of this research realm have been identified, and potential solutions addressing them have been proposed. On the whole, this review outlines a fascinating blueprint for the implementation of self-driven, high-performance, and highly integrated optoelectronic devices.

2. Strain-enabled homojunction photodetectors

As mentioned in the previous section, most self-driven 2DLM photodetectors have been constructed based on heterojunctions (*e.g.*, p-n junctions & Schottky junctions & homo-type junctions), where built-in electric field is formed on account of differentiated Fermi levels and photogenerated electron–hole pairs can be spontaneously separated with this impetus. However, 2DLM based heterojunctions commonly suffer from poor interface quality on account of a plethora of ingredients, such as loose interfacial contact by manual staking, serious interfacial contamination induced by transfer processing, and strong Fermi level pinning effects caused by interfacial defect states, making it quite challenging to achieve high-performance bias-free photodetection. Therefore, in recent years, researchers have been dedicated to exploring the transfer-free construction of 2DLM homojunctions. As is well known, the electronic band structure of low-dimensional materials is highly sensitive to strain.^{79–82} For example, density functional theory calculations have determined that the bandgap size of MoTe₂ reduces markedly from 1.04 to 0.8 eV as the strain increases from 0 to 4%.⁷⁹ In addition, first-principles calculations have found that both conduction band minimum and



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materials as well as their innovative application within the realm of optoelectronics.

valence band maximum of Bi_2S_3 monotonously downshift with increasing strain from 0 to 8%.⁸⁰ In this consideration, theoretically, the preparation of 2DLM homojunction zero-bias photodetectors can be achieved through rational strain engineering.

For this purpose, in 2022, Lu and collaborators prepared a strained $\text{MoS}_2/\text{MoS}_2$ homojunction self-driven photodetector by partially integrating monolayer MoS_2 on top of prefabricated Au nanoparticles (NPs), as schematically shown in Fig. 1a.⁷⁷ Within this device structure, a portion of the monolayer MoS_2 is shored up by the NPs underneath, leading to large biaxial tensile strain. As shown in Fig. 1b, under 660 nm illumination, pronounced photocurrent has been generated even at zero bias voltage. To further explore the physical mechanism of self-driven photodetection, first-principles calculations on the electronic band structures have been conducted (Fig. 1c). As the biaxial tensile strain increases, both the conduction band minimum (CBM) and valence band maximum (VBM) will downshift. As a consequence, a type-II band alignment has been formed between strained MoS_2 and pristine MoS_2 . In addition, the Fermi level largely upshifts with increasing strain. Therefore, a built-in electric field pointing from strained MoS_2 to pristine MoS_2 is generated, as evidenced by the KPFM measurement (Fig. 1d and e). Upon light excitation, electron-hole pairs are generated. Under the drive of the type-II band alignment and the built-in electric field, electrons are driven to the strained MoS_2 , and holes are driven

to the pristine MoS_2 , thus leading to the self-driven photo-response (Fig. 1f). On the whole, this study provides a facile route to realize bias-free homojunction photodetectors with seamless and contamination-free interfaces, which is conducive to suppressing the recombination and accelerating spacial separation of photogenerated electron-hole pairs. Furthermore, the electronic band structures of a wealth of 2DLMs, such as CrI_3 ,⁸³ CrS_2 ,⁸⁴ blue phosphorene,⁸⁵ ZrSe_3 ,⁸⁶ InSe ,⁸⁷ violet phosphorus,⁸⁸ *etc.*, have been revealed to be highly dependent on strain. In this consideration, the strain engineering strategy herein holds broad universality on constructing various self-driven 2DLM homojunction photodetectors, which are in need of further exploration in the upcoming future. Of note, a severe challenge for strain engineering is that the weak van der Waals force between 2DLM and substrate makes interlayer slippage easily occurs, resulting in inefficient strain transfer. To address this issue, coating the 2DLM channel with an encapsulation layer can be exploited, where the additional adhesion force supplied by the capping layer aids in stabilizing the 2DLM channel.⁸⁹

3. Homojunction photodetectors via thickness tailoring

As previously stated, the majority of self-driven 2DLM photodetectors have been constructed based on van der Waals het-

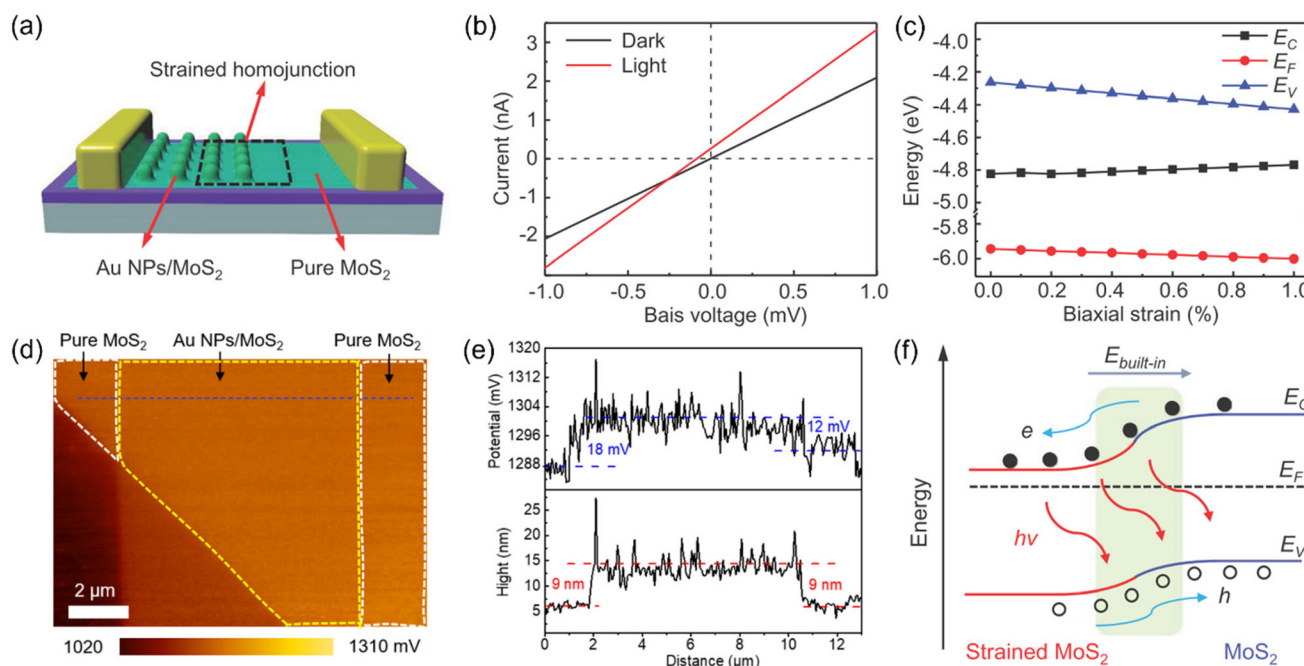


Fig. 1 (a) Schematic of the Au nanoparticles/ MoS_2 (strained MoS_2) and pristine MoS_2 in-plane homojunction self-driven photodetector. (b) I - V curves of the strained $\text{MoS}_2/\text{MoS}_2$ in-plane homojunction photodetector in darkness (black line) and under 660 nm illumination (red line). (c) Calculated conduction band minimum (E_C), valence band maximum (E_V), and Fermi level (E_F) of MoS_2 as a function of biaxial tension strain. (d) KPFM image of the in-plane homojunction. (e) Height (bottom panel) and surface potential (top panel) line profiles along the blue dotted line in (d). (f) Energy band diagram of the self-driven strained $\text{MoS}_2/\text{MoS}_2$ homojunction photodetector under illumination. Reproduced from ref. 77 with permission from John Wiley and Sons, Inc, copyright 2022.

erojunctions, intricate structures forged from two or more constituent components. Nevertheless, heterostructures have been commonly plagued by lattice mismatch, which engenders interfacial defects and distortions in the heterointerface. This, in turn, intensifies the trapping, scattering, and recombination of photocarriers, hindering the interfacial transport. Most notably, the traditional fabrication processes for preparing 2DLM-based heterojunctions, which frequently involve micro-mechanical exfoliation and transfer techniques, will inevitably introduce severe interfacial contamination, further degrading the device performances. These limitations significantly curtail the applicability of van der Waals heterojunctions in self-driven optoelectronic systems. In response to these tangible challenges, an ever-increasing number of researchers have turned their attention towards investigating and developing 2DLM homojunctions in a transfer-free manner, which feature seamless interfaces that function as ideal carrier transport pathways in optoelectronic devices.

Traditionally, 2DLM homojunctions have been prepared through localized doping.^{90,91} However, the implantation of exotic atoms can probably cause degeneration of crystal properties on account of the inconsistent atomic diameter and electronegativity of the doped atoms as compared to the substituted atoms. In addition, the invasive doping process may also bring about serious crystal defects spanning vacancies, antistites, and intercalated atoms. Contact doping is another potential strategy.^{92–94} Nevertheless, a substantial proportion of these dopants exhibit relatively poor stability. In this consideration, the long-term reliability of related homojunction devices still needs further improvement. Electrostatic doping, which is usually realized by using split gates,^{95–101} can avoid the adverse effects caused by heteroatom doping and the instability of surface modification doping, but this scheme entails additional voltage sources, which makes the devices suffer from large volume as the powered devices. Furthermore, the extra gate capacitance may crop the response. In addition to doping and electrostatic gating, thickness is also an important element affecting the electronic band structure of 2DLMs. For example, both density functional theory and GW calculations have unveiled that the work functions of transition-metal dichalcogenides have close relationship with the thickness.¹⁰² In addition, first-principles calculations indicate that the bandgap of AgInP_2S_6 monotonously increases with decreasing thickness.⁵² Therefore, the construction of 2DLM homojunctions *via* thickness tailoring warrants a new avenue.

As a proof, in 2024, Hu and coresearchers reported on the preparation of pristine MoTe_2 homojunction self-driven photodetectors *via* thickness tailoring.⁷⁸ Fig. 2a interprets the detailed preparation procedures. First, the molybdenum precursor nanofilms with asymmetric thicknesses are patterned on sapphire substrates using two-step photolithography and magnetron sputtering processes. Then, the MoTe_2 homojunctions are achieved *via* tellurization of the molybdenum patterns using a chemical vapor deposition method. As shown in Fig. 2b, under illumination, net positive photocurrent generates, demonstrating the feasibility for zero-bias photodetec-

tion. Importantly, the photoresponse is stable and highly reversible under periodic illuminations with various light power (Fig. 2c). In order to garner an in-depth understanding on the underlying working mechanism, DFT calculations have been performed. As shown in Fig. 2d, as the thickness increases, the bandgap of MoTe_2 decreases, leading to the shift of Fermi level. Consequently, a homojunction is formed between MoTe_2 layers with various thicknesses, with the built-in electric field pointing from thick- MoTe_2 to thin- MoTe_2 (Fig. 2e). When the homojunction is illuminated with light, electrons in the valence band are excited to the conduction band, generating electron-hole pairs. Then, under the drive of the built-in electric field, the photogenerated electrons transport from thin- MoTe_2 to thick- MoTe_2 , while photogenerated holes migrate from thick- MoTe_2 to thin- MoTe_2 , thereby resulting in the generation of photocurrent. Taking advantage of the superior scalability of the mature fabrication process for patterning metal pads, a 10×10 MoTe_2 homojunction photodetector array has been successfully constructed and harnessed for multi-pixel optoelectronic imaging (Fig. 2f). In conclusion, this finding spotlights a unique paradigm for the fabrication of bias-free 2DLM homojunction photodetectors precluding interface problems, thereby forging a path towards the realization of integrated and miniaturized photodetector systems featuring low energy consumption. By using the analogous working principle, a series of self-driven 2DLM homojunction photodetectors based on various 2DLMs, such as MoS_2 ,^{103,104} MoSe_2 ,^{105,106} WSe_2 ,¹⁰⁷ and ReS_2 ,¹⁰⁸ have been successfully constructed *via* thickness tailoring, underscoring the broad universality. Of note, most of the previous studies have been conducted based on micromechanical exfoliation technology, which, however, suffers from poor repeatability and controllability, making it impractical for practical application. Although the above two-step chemical vapor deposition method boasts commendable controllability, it falls short in terms of the interface quality of the resultant homojunction, exhibiting an unfavorable polycrystalline nature.⁷⁸ Fortunately, a promising two-step oxidation-etching methodology has demonstrated remarkable efficacy in achieving layer-by-layer thinning of 2DLMs with atomic-level precision.¹⁰⁹ Looking ahead, this technology can be further developed for constructing self-driven 2DLM homojunction photodetectors to achieve scalable and controllable customization of devices.

4. Homojunction photodetectors by structural engineering

In general, self-driven photodetection has been largely relied on the formation of built-in electric field, which usually exists at the heterojunctions, typically p-n junctions,¹¹⁰ Schottky junctions,^{111,112} as well as homo-type junctions (*i.e.* n-n and p-p structures).^{113–115} However, there exist a host of drawbacks for 2DLM based heterostructures, including interfacial defects/deformation brought by lattice mismatch, large interfacial gap brought by the weak interlayer adhesion, interfacial impurities

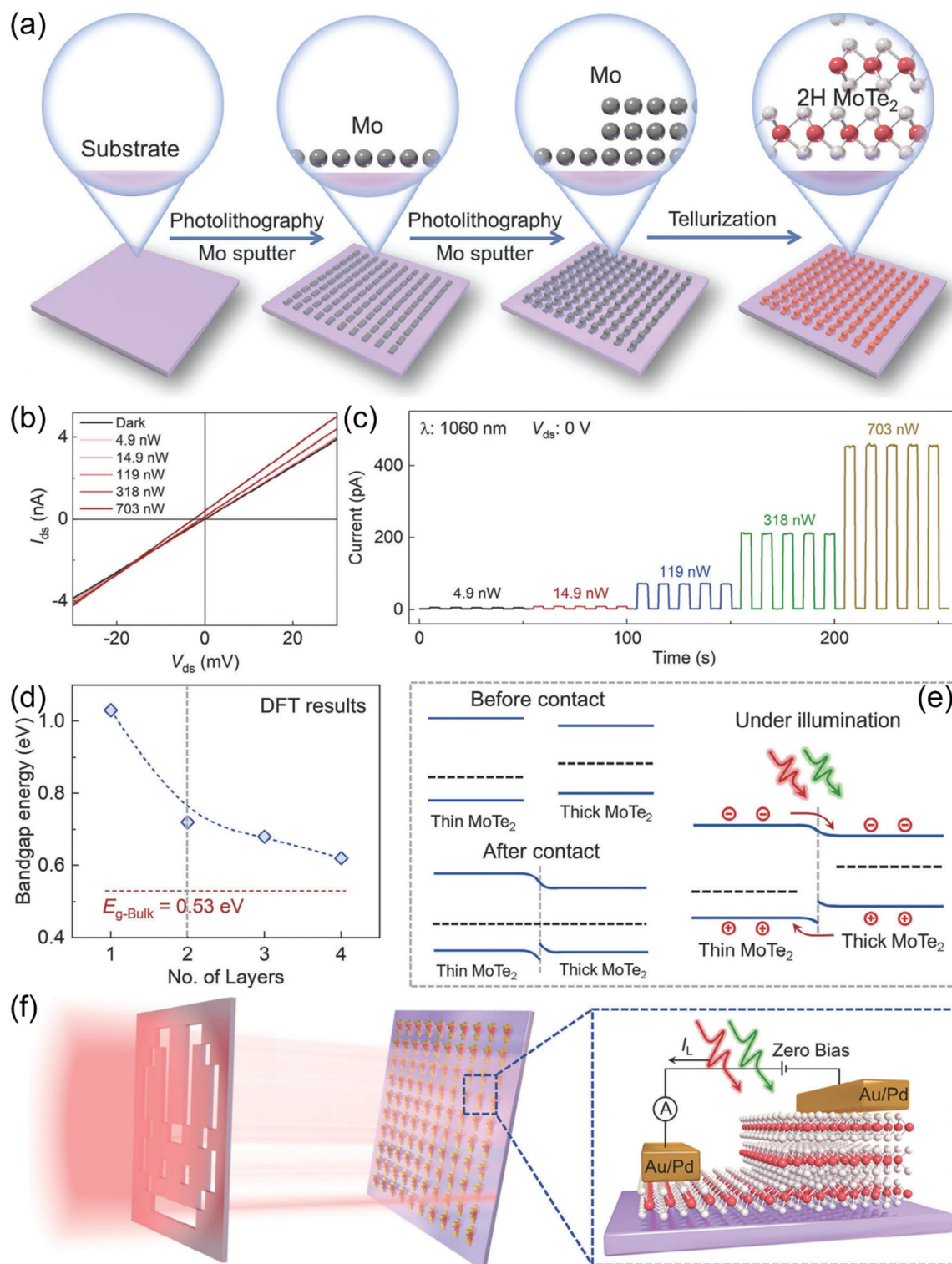


Fig. 2 (a) Schematic of the fabrication processes for preparing MoTe₂ homojunctions via thickness engineering. (b) Output curves of the MoTe₂ homojunction device in darkness and under 1060 nm light illumination with different light power. (c) Photoswitching curves under periodic 1060 nm illuminations with different light power under zero bias. (d) Bandgap as a function of layer number of MoTe₂ based on DFT calculations (blue diamonds and line). (e) Schematic diagrams of the energy band structures of thin-MoTe₂ and thick-MoTe₂ before and after contact (left panel) as well as the carrier dynamics of the thin-MoTe₂/thick-MoTe₂ homojunction upon light excitation (right panel). (f) Schematic illustration of the imaging process based on the MoTe₂ homojunction photodetector array. Reproduced from ref. 78 with permission from John Wiley and Sons, Inc, copyright 2024.

brought by mechanical exfoliation/transfer processing, *etc.* Despite that heterojunctions can be constructed by an *in situ* approach, epitaxial growth of van der Waals heterojunctions is quite challenging due to the high energy barrier required for nucleation and growth on the naturally passivated surface of 2DLMs compared with the conventional covalent materials.¹¹⁶ Within the context of the severe challenges in preparing heterostructures, the development of exceptional strategies aimed at the construction of built-in electric field is imperative and indispensable.

As a prototypical example, most recently, Zhang *et al.* have harnessed neoteric self-assembled 1D/2D WSe₂ homojunctions connected seamlessly by robust covalent bonds and exploited these structures for photodetection without external power.¹¹⁷ The detailed fabrication process has been schematically shown

in Fig. 3a, which contains the deposition and rolling-up steps. Firstly, 2D morphological WSe₂ nanosheets are prepared on SiO₂/Si substrates *via* a chemical vapor deposition (CVD) method. Then, volatile solution (typically alcohol) is dripped onto the substrate. The solution molecules can readily penetrate into the interspace between the WSe₂ nanosheets and the substrate and induce the dissociation. Following this, a spontaneous rolling-up process will occur under the drive of the surface tension.^{118,119} Of note, the area or degree of sample curling can be precisely controlled by the amount of alcohol solution, thus enabling the self-assembled construction of 1D/2D homostructures. As shown in Fig. 3b, first-principles calculations consolidate that the electronic band structures of 2D WSe₂ and 1D WSe₂ are highly distinct, which lays a solid foundation for self-driven photodetection. Fig. 3c showcases the

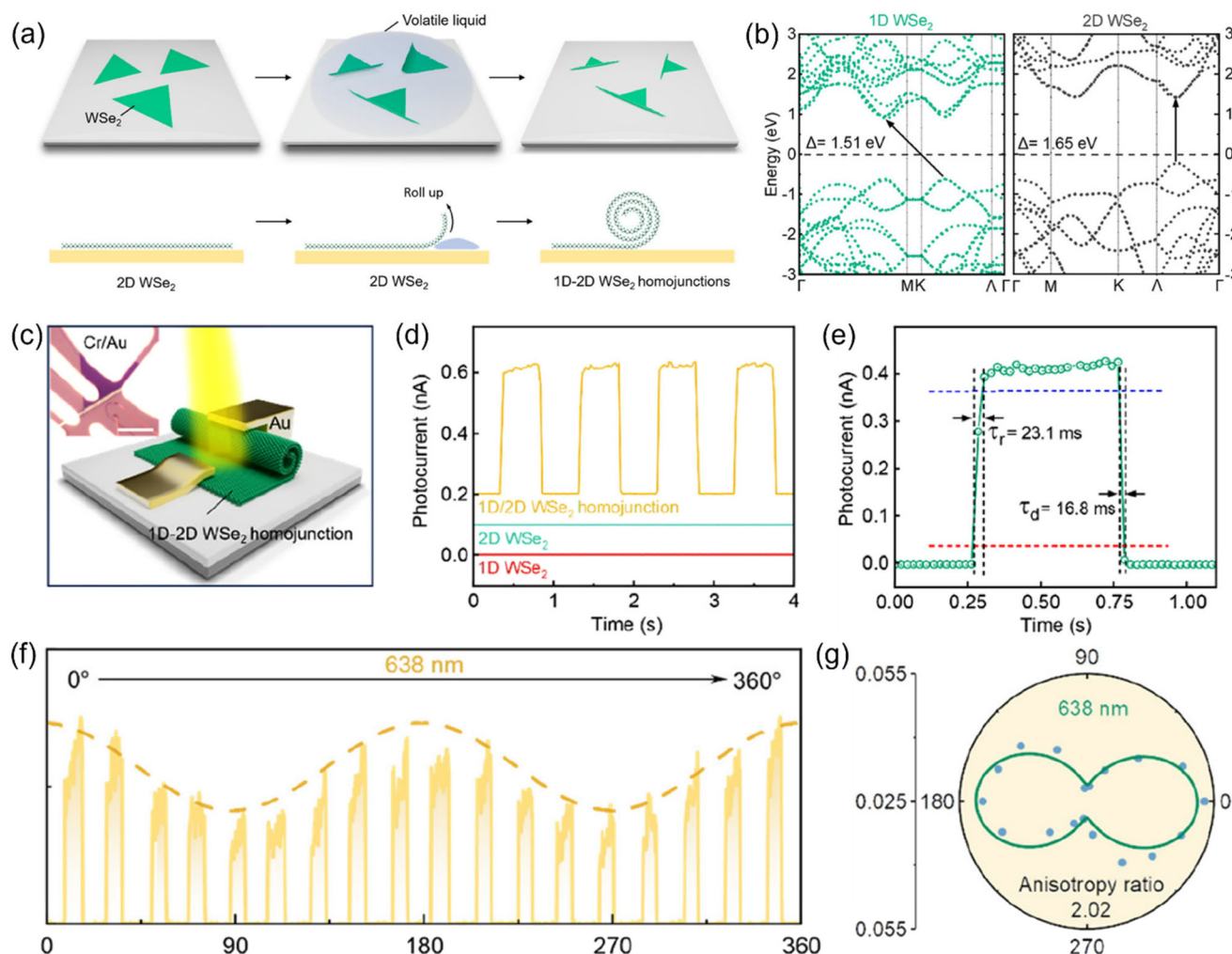


Fig. 3 (a) Schematic diagrams of the fabrication processes for constructing the 1D/2D WSe₂ homojunctions *via* a self-assembled strategy. (b) Band structures of 1D WSe₂ (left panel) and 2D WSe₂ (right panel) extracted from theoretical calculations. (c) Schematic diagram of a 1D/2D WSe₂ homojunction photodetector. The inset in the top-left corner shows an optical microscope image. (d) Photoswitching curves of the 1D WSe₂ (red line), 2D WSe₂ (green line), and 1D/2D WSe₂ homojunction (orange line) photodetectors under periodic illumination at zero bias. (e) The response time and decay time. (f) Anisotropic photoresponse of the 1D/2D WSe₂ homojunction device at zero bias under linearly polarized 638 nm illumination. (g) Polar plot of the photocurrent as a function of polarization angle, the angle between the 1D WSe₂ and the polarization direction of incident light. Reproduced from ref. 117 with permission from American Chemical Society, copyright 2024.

schematic diagram of a WSe₂ homojunction photodetector, where a heterogeneous interface is formed between 1D WSe₂ and 2D WSe₂. Specifically, this homojunction is tightly connected by robust covalent bonds rather than feeble vdW interactions of the traditional heterostructures, which avoids the excessive interfacial gap as well as the damage of defects, impurities and dangling bonds. Thus, the self-assembled 1D/2D homojunction offers an efficient interfacial charge transfer channel, which is conducive to high photosensitivity and speed. As depicted in Fig. 3d, under zero bias condition, the 1D/2D homojunction photodetector exhibits a pronounced photoresponse to periodic 660 nm illumination, in stark contrast to the negligible photoresponse displayed by both the 1D WSe₂ (curling degree of $\approx 100\%$) and 2D WSe₂ photodetectors (curling degree of $\approx 0\%$). These results suggest that the self-driven characteristics are induced by the unique composite 1D/2D homojunction structure. According to the DFT calculations, the Fermi level of 1D WSe₂ is higher than that of 2D WSe₂. As a consequence, a built-in electric field pointing from the former to the latter is formed after these components are in contact. Upon light excitation, photocarriers are generated, after which electrons drift to the 1D WSe₂ while holes drift to the 2D WSe₂ under the drive of the built-in electric field, thereby resulting in the self-driven photoresponse. As shown in Fig. 3e, the WSe₂ homojunction device possesses fast response rate, with the rise and decay time reaching 23.1 ms and 16.8 ms, respectively. Most importantly, by virtue of the strong anisotropy of 1D WSe₂, the homojunction device exhibits pronounced polarized photosensitivity. As clearly shown in Fig. 3f and g, the photocurrent exhibits approximate periodic variation with changing polarization angle and reaches the maximum/minimum magnitude at 0°/90°. Specifically, a dichroic ratio up to ≈ 2.02 has been achieved under 638 nm illumination, which demonstrates a competitive polarization sensitivity. In general, the study depicts an innovative protocol towards homojunction photodetectors featuring intact interface as well as exceptional functionality, opening up new opportunities for highly integrated and multifunctional photoelectric detection.

5. Homojunction photodetectors induced by asymmetric nonvolatile ferroelectric gating

Due to the atomic-level channel thickness, the electrostatic screening ability of 2DLMs is extremely feeble. In this consideration, the polarity of charge carriers of 2DLMs can be flexibly controlled by the electrostatic field of the gate. On this basis, the preparation of 2DLM p–n homojunctions can be achieved through localized regulation of the gate voltage, providing an efficient strategy for realizing zero-bias photoelectric detection. In reality, through electrostatic gating, self-driven photodetectors based on a series of 2DLMs, including WSe₂,^{95–97} MoSe₂,⁹⁸ black phosphorus,⁹⁹ and MoTe₂,¹⁰⁰ have

been successfully produced. Nevertheless, the operation of these devices necessitates the sustained application of gate voltage, which entails additional voltage sources, making the volume of the devices too large to achieve compact integration. Therefore, researchers have been committed to exploring alternative strategies for achieving self-driven photodetection. Ferroelectric materials, due to the non-volatile polarization, can induce electrostatic field in the adjacent regions, thereby providing promising material platforms for the construction of 2DLM p–n homojunctions without gate voltage sources.¹²⁰

As a demonstration, in 2020, Wu *et al.* achieved self-driven photodetection by integrating a MoTe₂ nanosheet onto a ferroelectric poly(vinylidene fluoride-trifluoroethylene) (P(VDF-TrFE)) dielectric layer with opposite polarization directions in coterminous regions (Fig. 4a).¹²¹ As shown in Fig. 4b, the MoTe₂ device integrated on the asymmetrically polarized ferroelectric dielectric layer exhibits remarkable rectification characteristics in darkness, which signifies the existence of a significant interfacial potential barrier, underscoring the formation of an excellent p–n junction. Under illumination, the device exhibits distinct photocurrent even at a source–drain voltage of 0 V. Given the exceedingly low dark current under a self-driven operating mode, the MoTe₂ device achieves an impressively high on/off ratio, attaining a value nearing $\approx 10^6$ when illuminated with a light power of 300 μ W (Fig. 4c). Meanwhile, the response time and recovery time are as short as 30 and 45 μ s, respectively (Fig. 4d and e). The scanning photocurrent mapping measurement reveals that the most pronounced photoresponse appears in the interfacial region where the polarities of the coterminous ferroelectric dielectric layer are opposite (Fig. 4f), indicating that the photovoltaic characteristics of the device is predominantly emanated from the formation of a p–n homojunction. Specifically, when the ferroelectric dielectric layer is polarized downward, holes are accumulated in MoTe₂, leading to the formation of p-type MoTe₂. By contrast, n-type MoTe₂ is formed when the ferroelectric dielectric layer is polarized upward. Thus, by reversing the polarity of coterminous regions of the ferroelectric dielectric layer, the preparation of MoTe₂ p–n homojunctions can be achieved. Under illumination, photocarriers are generated. Driven by the built-in electric field, electrons and holes are transported in opposite directions, thereby achieving bias-free photodetection. Analogously, Lv and colleagues showcased a self-driven MoS₂ photodetector, achieved by integrating a MoS₂ nanosheet with a P(VDF-TrFE) ferroelectric layer featuring reversed polarization directions in contiguous regions.¹²² Following this success, Zhang *et al.* determined that the photosensitivity of the ferroelectricity-enabled self-driven MoS₂ photodetector can be boosted by several times with the modification of a plasmonic Au array, which can be associated with the strong confinement of free space light to the vicinity of the light-sensing channel.¹²³ In another study, Wu *et al.* achieved a self-driven polarization-sensitive BP photodetector by exploiting the same working mechanism.¹²⁴ Specifically, the dichroic ratios reach 3.9 and 288 upon 520 and 1450 nm illuminations, respectively. Following the implementation of single-junction

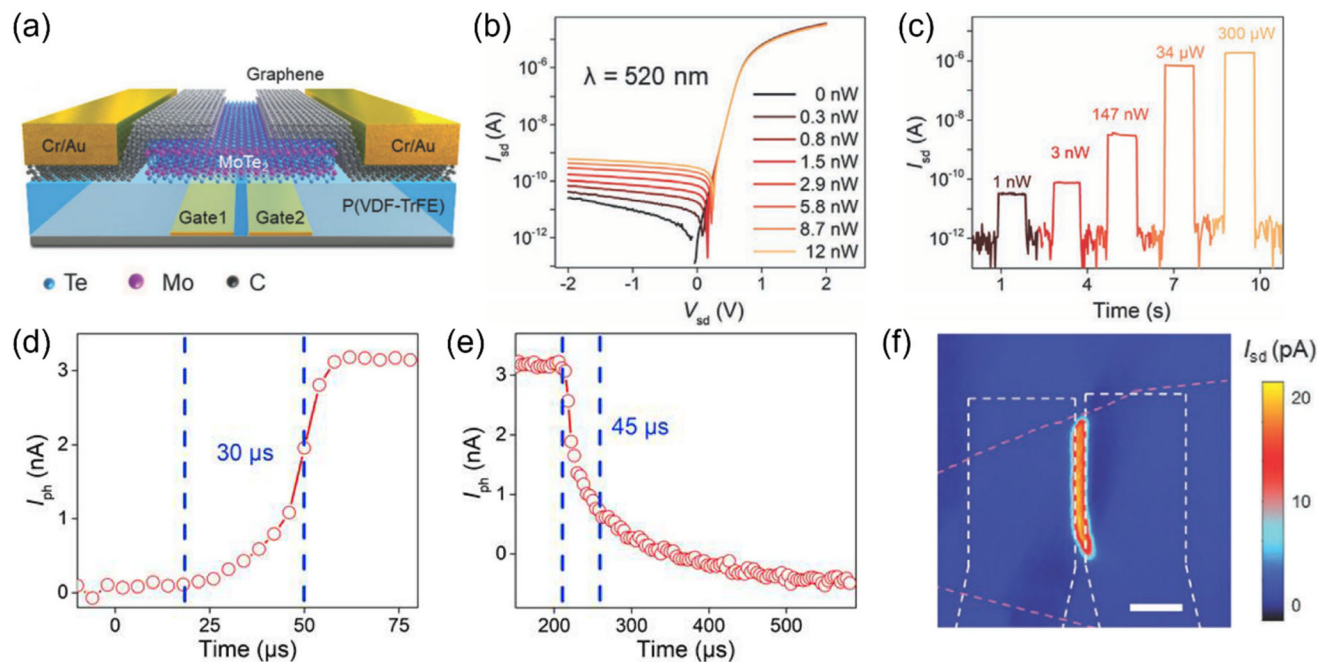


Fig. 4 (a) Schematic illustration of a self-driven MoTe₂ photodetector enabled by ferroelectric gate. (b) I_{sd} - V_{sd} curves in darkness and under 520 nm illuminations with various light power. (c) Photoswitching curves under illuminations with various light power at zero bias. (d–e) Rising and decaying edges of the temporal photoresponse. (f) Photocurrent mapping image under a source–drain bias of 0 V. Reproduced from ref. 121 with permission from John Wiley and Sons, Inc., copyright 2020.

devices, by connecting the ferroelectricity-enabled MoTe₂ p–n homojunctions in series, Wu *et al.* demonstrated the linear amplification of the output photovoltage, providing an adaptable route for signal enhancement.¹²⁵ In summary, these studies present an innovative and straightforward approach for realizing high-performance self-driven photodetectors that boast a seamless interface and eliminate the need for sustained application of external gate voltage by leveraging ferroelectric dielectrics as an alternative. It is worth emphasizing that polymer ferroelectrics, exemplified by P(VDF-TrFE), have predominantly been utilized in the past research endeavors on account of the economic viability and robust polarization intensity. However, due to the use of multiple organics in the liftoff process, organic ferroelectrics are incompatible with the currently well-established microfabrication processes, which is harmful to the scalability and miniaturization. Therefore, inorganic ferroelectric dielectrics immune to organic solvents need to be developed as alternatives in the future. Recently, a range of 2DLMs have been documented to exhibit intrinsic ferroelectricity,^{126,127} providing new material platforms for the next-generation ferroelectricity-enabled self-driven 2DLM photodetectors.

6. Asymmetric contacts

6.1 Asymmetric contacts with differentiated work functions

Schottky junction is a kind of heterojunction formed by the contact between a metal and a semiconductor with differen-

tiated work functions. In principle, when the work function of a metal is larger than that of an n-type semiconductor or smaller than that of a p-type semiconductor, theoretically a Schottky junction can be formed. Ideally, in the well-established Schottky–Mott rule, for n-type semiconductors, the Schottky barrier height (Φ_{SB}) is extracted by

$$\Phi_{SB} = \Phi_M - \Phi_S, \quad (1)$$

where Φ_M is the work function of the metal, and Φ_S is the work function of the semiconductor. By contrast, the Schottky barrier height for p-type semiconductors is extracted by

$$\Phi_{SB} = \Phi_S - \Phi_M. \quad (2)$$

Analogous to p–n junctions, built-in electric field and depletion layer will be formed at the heterointerface in vicinity of Schottky junctions, where photocarriers can be efficiently separated. In this regard, Schottky junctions can also be applied to bias-free photodetection. It is noteworthy that the fabrication of Schottky junctions necessitates no intricate doping procedures, thereby significantly simplifying the production process in comparison to that of p–n junctions. Nonetheless, compared with traditional bulk semiconductors, 2DLM have encountered tough challenges in constructing vertical Schottky junctions, the common protocols for bulk semiconductors, due to its extremely thin thickness, which makes the top and bottom electrodes easily misconnect. In this consideration, there is an urgent demand to develop new configurations of self-driven Schottky junction photodetectors. In

recent years, the lateral MSM structure has reawakened the attention of researchers. In a typical MSM structure, symmetric electrodes will be adapted. Therefore, the effects of the coupled Schottky junctions of the source and drain electrodes counteract each other. Consequently, net photoelectric effect cannot be generated without external bias. To address this issue, asymmetric MSM structures have been proposed for zero-bias photodetectors. Compared with p-n junction and vertical Schottky junction self-driven photodetectors, self-driven photodetectors based on lateral asymmetric MSM structures have the following advantages. Firstly, unlike p-n junctions, the construction of Schottky junctions is relatively simple and free from the elaborate doping of semiconductor materials, which can avoid performance degradation caused by doping. Secondly, compared to the vertical structures, lateral structures are more compatible with modern planar microfabrication

techniques and thereby are more conducive to the fabrication of device arrays. Moreover, the in-plane transport property of 2DLM is generally more efficient than the out-of-plane one, since the latter suffers from relatively large interplanar spacing. Consequently, asymmetric lateral MSM structures have considerable potential for application in bias-free light sensing.

As proof of their potency, in 2022, Han *et al.* demonstrated a self-driven $\text{Bi}_2\text{O}_2\text{Se}$ photodetector by simultaneously using Pd and Ti as contacts (Fig. 5a).¹²⁸ As shown in Fig. 5b, the work functions of Pd, Ti, and $\text{Bi}_2\text{O}_2\text{Se}$ are 5.12, 4.26, and 4.36 eV, respectively. Accordingly, Pd forms a Schottky junction with $\text{Bi}_2\text{O}_2\text{Se}$, while Ti forms an Ohmic contact with $\text{Bi}_2\text{O}_2\text{Se}$. As a consequence, a distinct rectification effect has been realized (Fig. 5c). As shown in Fig. 5d, under zero bias, the Pd/ $\text{Bi}_2\text{O}_2\text{Se}$ /Ti device exhibits distinct photoswitching character-

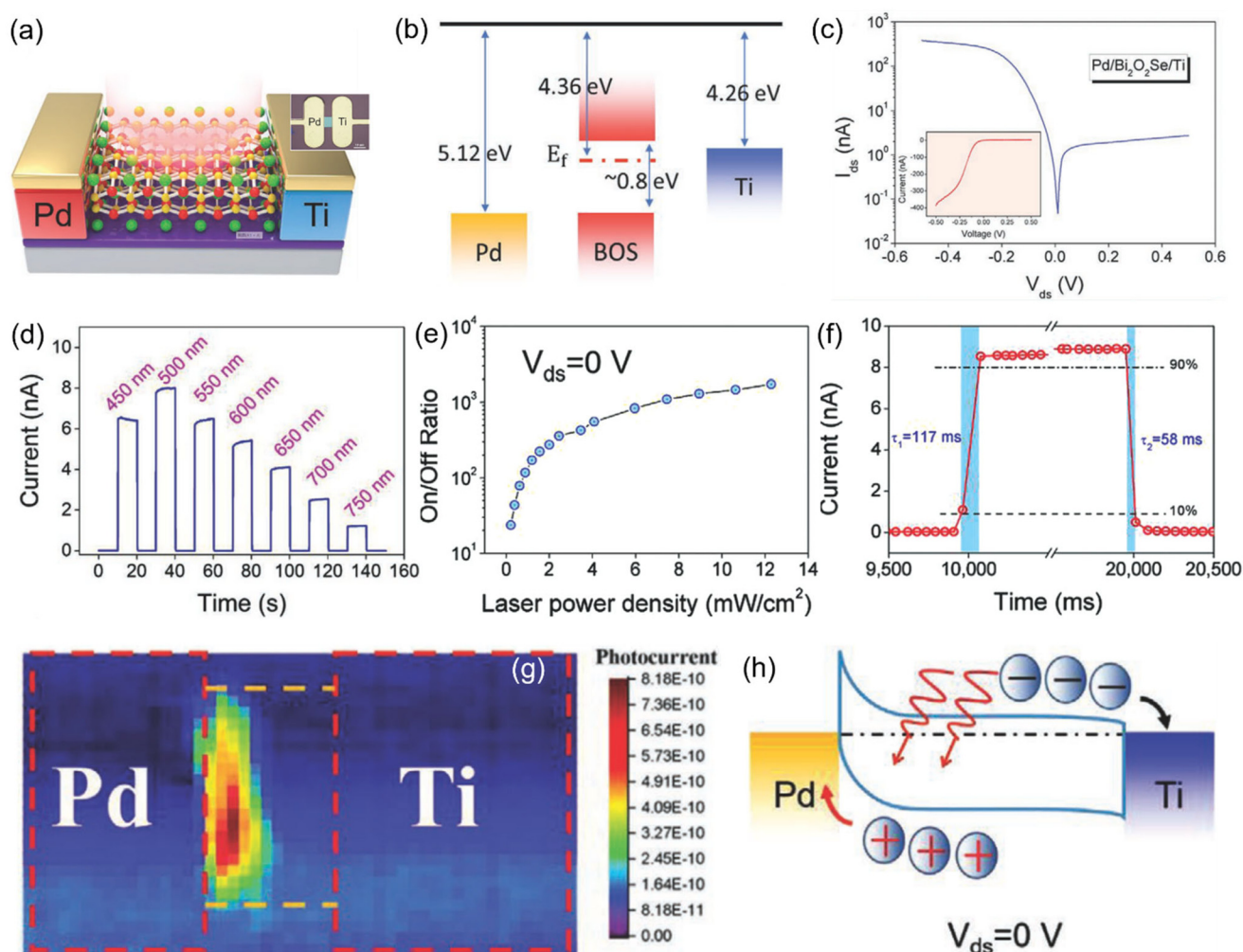


Fig. 5 (a) Schematic diagram of a lateral Pd/ $\text{Bi}_2\text{O}_2\text{Se}$ /Ti photodetector. The inset presents the corresponding optical microscopy image. (b) The energy band diagrams of Pd, Ti, and $\text{Bi}_2\text{O}_2\text{Se}$ before contact. (c) I - V curve in darkness in logarithmic scale. The inset presents the I - V curve in linear scale. (d) Typical photoswitching curves upon illuminations with various wavelengths under zero bias. (e) On/off ratio as a function of light power density. (f) Temporal photoresponse. (g) Photocurrent mapping image of the Pd/ $\text{Bi}_2\text{O}_2\text{Se}$ /Ti photodetector under zero bias. (h) Energy band diagram of the device without external bias under irradiation. Reproduced from ref. 128 with permission from John Wiley and Sons, Inc, copyright 2022.

istics upon light illuminations with various wavelengths. By virtue of the low dark current without external bias, the device exhibits high on/off ratio, which increases with increasing light intensity (Fig. 5e). Specifically, under 500 nm illumination with a light intensity of 12.4 mW cm^{-2} , the on/off ratio approaches ≈ 1700 . In addition, the Pd/Bi₂O₃Se/Ti device exhibits fast response rate with the rise/decay time down to 117/58.5 ms (Fig. 5f), which is benefit from the high in-plane carrier mobility of Bi₂O₃Se. The working principle of the self-driven Pd/Bi₂O₃Se/Ti photodetector can be intuitively unveiled in the photocurrent mapping measurement (Fig. 5g), which evidently determines that the photoresponse is largely originated from the Pd/Bi₂O₃Se heterointerface. Therefore, the zero-bias photoresponse can be well explained as follows (Fig. 5h). Upon light illumination, electron–hole pairs are generated by interband excitation. Since Pd and Bi₂O₃Se form a Schottky junction with the built-in electric field pointing from Bi₂O₃Se to Pd, under the drive of the built-in electric field, photogenerated electrons in vicinity of the Pd/Bi₂O₃Se heterogeneous interface will flow to the Ti electrode, while photogenerated holes will flow to the Pd electrode. By contrast, the photovoltaic effect of the Ti/Bi₂O₃Se junction is negligible due to the similar Fermi levels. Accordingly, net photocurrent flowing from Pd electrode to Ti electrode is generated. On the whole, this study depicts a general avenue for realizing self-driven 2DLM photodetectors by Fermi level nonequivalence of source and drain contacts. Up to now, a variety of self-driven 2DLM photodetectors enabled by asymmetric electrodes with differentiated work functions, including Pd/graphene/Ti,¹²⁹ Ag/MoSe₂/Pt,¹³⁰ Ag/MoSe₂/Pt,¹²⁰ Au/InSe/In,¹³¹ Au/MoSe₂/Ti,¹³² Au/MoSe₂/Ag,¹³³ Pd/WSe₂/Ti,¹³⁴ Au/MoSe₂/Mo₂C,¹³⁵ Au/WS₂/Ti,¹³⁶ Ag/MoSe₂/MoO₃,¹³⁷ Ag/Bi₂O₃Se/Ni,¹³⁸ Au/h-BN/Ti,¹³⁹ Au/SbI₃/Pt,¹⁴⁰ and Pt/MoSe₂/Ti,¹⁴¹ have been realized.

As mentioned above, designing Schottky junctions by interfacing 2DLMs with metallic materials is a crucial approach for achieving high-performance self-driven photodetectors. Ideally, Schottky–Mott rule helps in predicting and adjusting the electronic properties of metal–semiconductor junctions, which is significant for designing and optimizing Schottky junction photodetectors. However, this framework is often invalid for the contact between 3D metallic materials and 2DLMs in practical situations. Since there is no direct chemical bonding between 3D metals and 2DLMs, the resulting atomic disorders often lead to a large number of defect states which are difficult to avoid. Accordingly, although there are distinct differences in the work functions of various metals, the large number of defect states at the contact interface between 3D metals and 2DLMs leads to severe Fermi level pinning, resulting in a relatively limited built-in electric field and thereby a narrow depletion region width (W) given by:

$$W = \sqrt{\frac{2\epsilon_0\epsilon_r\phi_B}{eN_s}}, \quad (3)$$

where ϵ_0 is the permittivity of free space, ϵ_r is the relative dielectric constant of the semiconductor, ϕ_B is the barrier

height, e is the elementary charge, N_s is the carrier density of the semiconductor. To address this predicament, a series of 2D metals/semimetals such as graphene, 1T'-MoTe₂, and WTe₂ have been harnessed as high-performance contact electrodes, significantly improving the performance of 2DLM-based photodetectors. Due to the absence of interfacial dangling bonds, the interfacial defect state density between 2DLMs is much lower as compared to that of interface between 3D metals/semimetals and 2DLMs. On account of the low density of defect states, which helps alleviate the Fermi level pinning effect, the interfacial band alignment can be realized more easily at the interface between 2D semimetal and 2D semiconductor.

As a demonstration, in 2024, Wang and collaborators have developed the asymmetric 1T'-MoTe₂/WS₂/Au Schottky junctions for self-driven photodetection (Fig. 6a).¹⁴² As shown in Fig. 6b, compared to the conventional interface between 3D metals and 2D semiconductors severely degraded by Fermi level pinning, the 2D metal/2D semiconductor interface exhibits a much higher as-designed Schottky barrier. As shown in Fig. 6c and d, the asymmetric Schottky structure exhibits notable photocurrent at $V_{ds} = 0 \text{ V}$. The underlying working mechanism for the self-driven characteristics has been schematically depicted in Fig. 6e. On account of the severe Fermi level pinning effect, the built-in electric field at the WS₂/Au interface is substantially smaller than the theoretical value. By contrast, taking advantage of the intact van der Waals interface, built-in electric field at the 1T'-MoTe₂/WS₂ junction is much stronger. As a consequence, the balance of the two electrode/channel interfaces is broken. Upon zero bias, photogenerated carriers are spontaneously separated, and a net photocurrent is generated due to the asymmetric interfacial built-in electric field. Overall, the use of 2D metals/semimetals as electrodes greatly enriches the available categories of Schottky junctions, providing a wide range of building blocks for constructing various types of high-performance self-driven 2DLM based Schottky junction photodetectors. Thus far, self-driven Gr/WSe₂/Au,¹⁴³ 1T'-MoTe₂/MoSe₂/Cr,¹⁴⁴ Gr/WS₂/Pt,¹⁴⁵ 1T'-MoTe₂/InSe/Cr,¹⁴⁶ Pd/GeSe/Gr,¹⁴⁷ PtSe₂/WSe₂/Au,¹⁴⁸ Gr/WSe₂/Au,¹⁴⁹ Cr/BP/WTe₂,¹⁵⁰ Gr/WSe₂/Au,¹⁵¹ Sb₂Te₃/WSe₂/Au,¹⁵² and Gr/WS₂/Au¹⁵³ photodetectors have been successfully demonstrated.

However, in practical applications, this strategy also has its own limitations. Specifically, the transfer process commonly involved in the fabrication of the heterogeneous electrode/channel van der Waals structures is prone to induce extrinsic contamination to the heterointerface, leading to a degradation in the contact quality. In addition, there generally exists a relatively large van der Waals gap at the heterogeneous interface of transferred van der Waals heterojunctions, which is also adverse to interfacial carrier transport. To resolve these issues, in 2023, by exploiting the thickness-dependent electronic band structure of 2DLMs, Zhang and collaborators developed a self-driven PdSe₂ photodetector by exploiting conventional covalent metal contact and *in situ* integrated PdSe₂ contact.¹⁵⁴ Specifically, a thickness-uneven PdSe₂ nanosheet is prepared

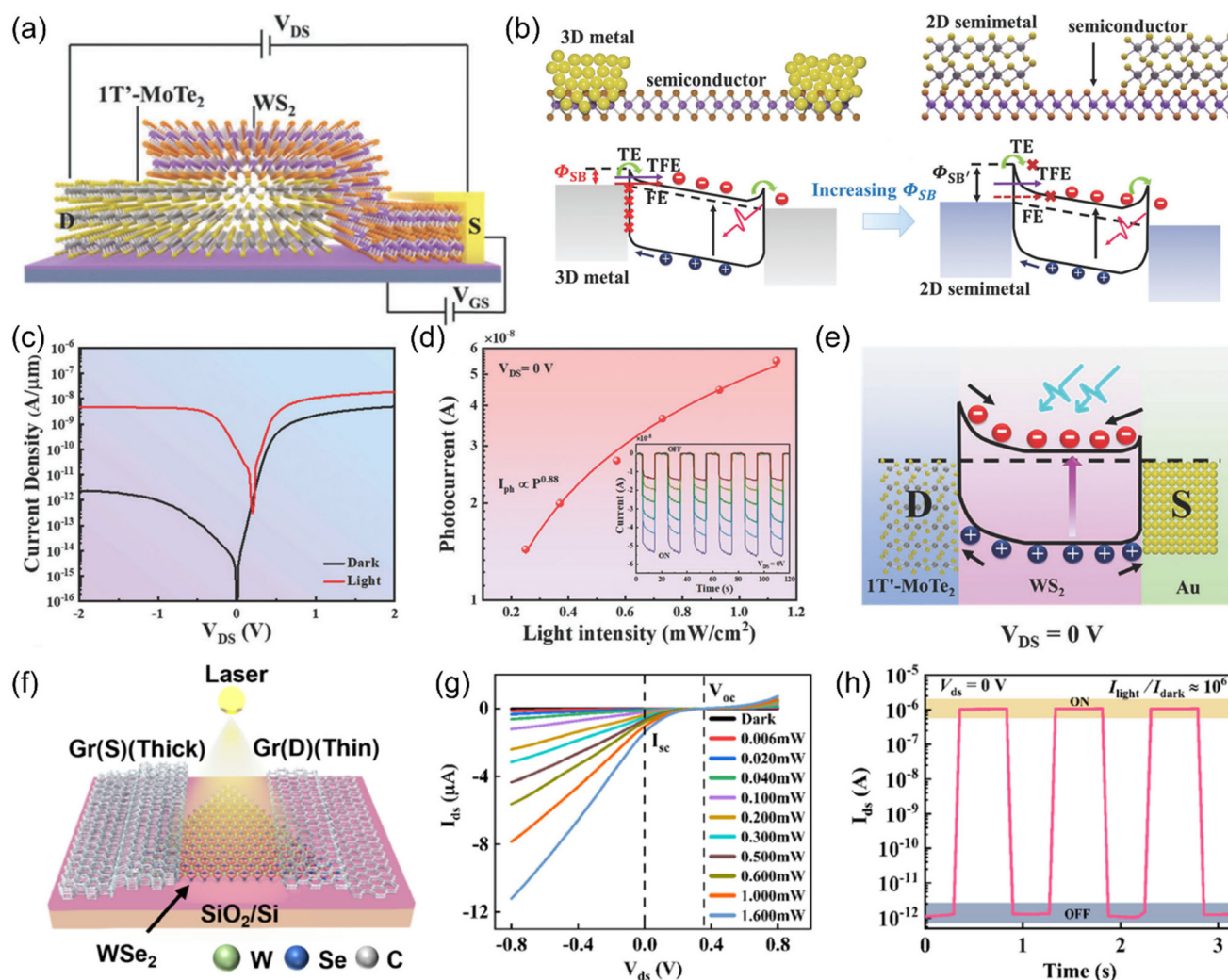


Fig. 6 (a) Schematic of a self-driven 1T'-MoTe₂/WS₂/Au photodetector. (b) Schematic diagrams of common 3D metal/semiconductor (left) and ideal 2D semimetal (metal)/semiconductor (right) Schottky structures (top panel) and the corresponding band structures (bottom panel). (c) *J*-*V* curves under dark (black line) and light illumination (red line). (d) Photocurrent as a function of light intensity. The inset summarizes the photo-switching curves. (e) Energy band diagram depicting the dynamic process of photocarriers at $V_{ds} = 0$ V. Reproduced from ref. 142 with permission from John Wiley and Sons, Inc., copyright 2024. (f) Schematic illustration of a self-driven MSM-type WSe₂ photodetector with graphene electrodes of different thicknesses. (g) *I*-*V* curves in darkness and under illuminations with various light intensities. (h) Photoswitching curve. Reproduced from ref. 155 with permission from American Chemical Society, copyright 2023.

by mechanical exfoliation and then transferred onto a substrate with pre-patterned Au electrodes. The Au pad contacting directly to the thin PdSe₂ layer serves as one of the electrode. Since the thick PdSe₂ layer is metallic, it serves as the other electrode for the thin PdSe₂ channel, featuring damage-free, clean and seamless electrode/channel interface. Consequently, an asymmetric MSM-type photodetector has been successfully constructed. At zero bias, photocurrent mapping measurement reveals that opposite photocurrents appear at the contact region of the thin PdSe₂/Au and the homojunction region between the thin and the thick PdSe₂ nanoflakes, which jointly contribute to the self-driven photoresponse. In addition, the device exhibits fast response speed. The response time and recovery time are as short as 0.24 and 0.72 ms,

respectively. As a whole, this study exemplifies a straightforward approach for achieving high-performance and contamination-free self-driven 2DLM photodetectors by thickness engineering.

In addition, to further mitigate the detrimental interface effects, both electrodes can be replaced with 2D layered metals/semimetals. As an example, in 2023, Tong and collaborators prepared a self-driven Gr/WS₂/Gr photodetector based on asymmetric van der Waals contacts (Fig. 6f).¹⁵⁵ Within this structure, graphene electrodes with different thicknesses have been exploited. Fig. 6g presents the *I*-*V* curve in darkness and under light illumination. The significant rectifying effect substantiates the formation of asymmetric interfacial barriers. In addition, the *I*-*V* curves detour the origin under illumination,

demonstrating the occurrence of zero-bias photocurrent. Benefiting from the low dark current and highly efficient carrier collection, distinct on/off switching with the on/off ratio reaching $\approx 10^6$ has been achieved (Fig. 6h). The underlying mechanism can be explained as follows. Since the electronic band structure of graphene is closely related with thickness, the employment of asymmetric graphene electrodes results in the difference in the interfacial built-in electric field. As a consequence, the balance of the two electrode/channel interface is broken, leading to the self-driven photoresponse. Most recently, Wang *et al.* reported on a zero-bias WS_2 photodetector by using TaSe_2 electrodes with different thicknesses.¹⁵⁶ In addition to thickness tailoring, heteroatom doping is another approach to modulate the electronic structure of 2D electrodes. For a prototypical example, in 2022, Guo *et al.* developed a self-driven MoS_2 photodetector by exploiting pristine graphene and N-doped graphene as electrodes.¹⁵⁷ Under zero-bias voltage, the optimized responsivity and detectivity reach 0.63 A W^{-1} and 7.71×10^{12} Jones, respectively. Moreover, the symmetry of the device can also be broken by using two distinct van der Waals semimetals/metals. Most recently, by integrating graphene and NbSe_2 electrodes within a WSe_2 photodetector, He *et al.* have achieved a self-driven photodetector with an ultra-high on/off ratio exceeding 10^5 .¹⁵⁸

It is worth noting that the above research works have mostly been based on the preparation of lateral MSM structures. However, there exist two potential shortcomings in terms of the structure of lateral MSM-type photodetectors. On one hand, the heterointerface of the lateral electrode/channel contact is substantially limited (*i.e.* quasi one-dimensional geometry in theory). This will lead to a substantially narrow depletion layer region, which is adverse to the spacial dissociation of photogenerated electron-hole pairs. On the other hand, due to the resolution limitation of micromachining processing technology, it is challenging to shorten the spacing between electrodes of the lateral devices (normally $> 2 \mu\text{m}$ for photolithography), which makes the transport path of photocarriers relatively long. This is adverse to the high-efficiency collection of photocarriers. Therefore, there is still an imperative demand to develop self-driven MSM-type photodetectors with vertical structures manifesting large contact area (*i.e.* 2D depletion layer), nanoscale transport path, *etc.* However, in terms of constructing vertical MSM structures, traditional electrodes have exhibited a host of shortcomings, including numerous surface dangling bonds, poor flatness, and low light transmittance. In recent years, a wealth of semi-metallic/metallic 2DLMs have been harnessed for use as electrodes, including graphene,¹⁵⁹ $1\text{T}'\text{-WTe}_2$,¹⁶⁰ Bi_2Te_3 ,¹⁶¹ CoS_2 ,¹⁶² PtTe_2 ,¹⁶³ PtBi_2 ,¹⁶⁴ *etc.* These van der Waals materials typically have atomically flat surfaces, which are beneficial for suppressing notorious short-circuiting effect caused by surface undulations. In addition, the dangling-bond-free surface of 2DLMs can effectively suppress interfacial strain and defects, thereby reducing interfacial scattering and recombination of charge carriers. Furthermore, by taking advantage of the atomic-level thickness, these electrodes have relatively high light transmit-

tance, allowing more incident light to pass through when used as top electrodes. By virtue of the above advantages, 2DLMs have indisputable potential for application in vertical self-driven MSM-type photodetectors.

As an example, in 2019, Wei *et al.* constructed asymmetric vertical $\text{Gr}/\text{MoTe}_2/\text{Gr}$ heterojunctions and successfully exploited them for self-driven photodetection (Fig. 7a).¹⁶⁵ The asymmetry is mainly due to the distinct material properties enabled by different preparation methods. Specifically, p-type monolayer graphene layers grown by chemical vapor deposition (CVD) were used as the top contacts, n-type exfoliated multiple graphene layers were used as the bottom contacts, and exfoliated multilayer MoTe_2 layers were used as the light-sensing channels. In dark, the $\text{Gr}/\text{MoTe}_2/\text{Gr}$ device exhibits distinct rectification effect, which is associated with the asymmetric interfacial barriers. Upon 1064 nm illumination, the I - V curve evidently bypasses the origin, indicative of the photovoltaic effect, which lays a solid foundation for self-driven photodetection. With the advantage of the atomic-scale transport path (*i.e.* the ultrathin thickness of the MoTe_2 channel, $\approx 15 \text{ nm}$ in this case), the vertical device exhibits a fast response rate with the rise/decay time down to 16.58/14.96 μs (Fig. 7b). Photovoltage mapping measurement reveals that the photovoltage is predominantly generated at the vertically overlapping region of the three flakes, which indicates that the photoresponse is originated from the heterointerface. Accordingly, the physical mechanism can be explained as follows. As shown in Fig. 7c, due to the difference in the Fermi levels of various components, built-in electric fields will be formed from n-type graphene to MoTe_2 and from MoTe_2 to p-type graphene after contact and equilibrium. Upon light excitation, photocarriers are generated, after which electrons will be driven to the n-type graphene and holes will be driven to the p-type graphene. As a consequence, a current flow from n-type graphene to p-type graphene is generated. Furthermore, it is further revealed that the thickness of the MoTe_2 channel plays a critical role in the device properties. As shown in Fig. 7d, the photoresponse generally increases with increasing channel thickness. This is easily understandable since the light absorption enhances with increasing channel thickness. On the whole, the study has put forward a new scheme toward high-speed and high-sensitivity self-driven photodetection.

6.2 Asymmetric contacts with distinct contact lengths

The integration of asymmetric electrodes with different work functions within one device has proven as an effective method for constructing MSM-type self-driven 2DLM photodetectors.^{128,131,141} In spite of the significant progress, this strategy has also come with its own formidable challenges. On one hand, a careful configuration of the work functions of the electrodes is required. On the other hand, due to the necessity to integrate two different types of metal electrodes, the fabrication process for preparing these asymmetric devices becomes exceptionally complex, requiring the deposition of the first kind of electrode, followed by the precise positioning and deposition of the second kind of electrode by using high-

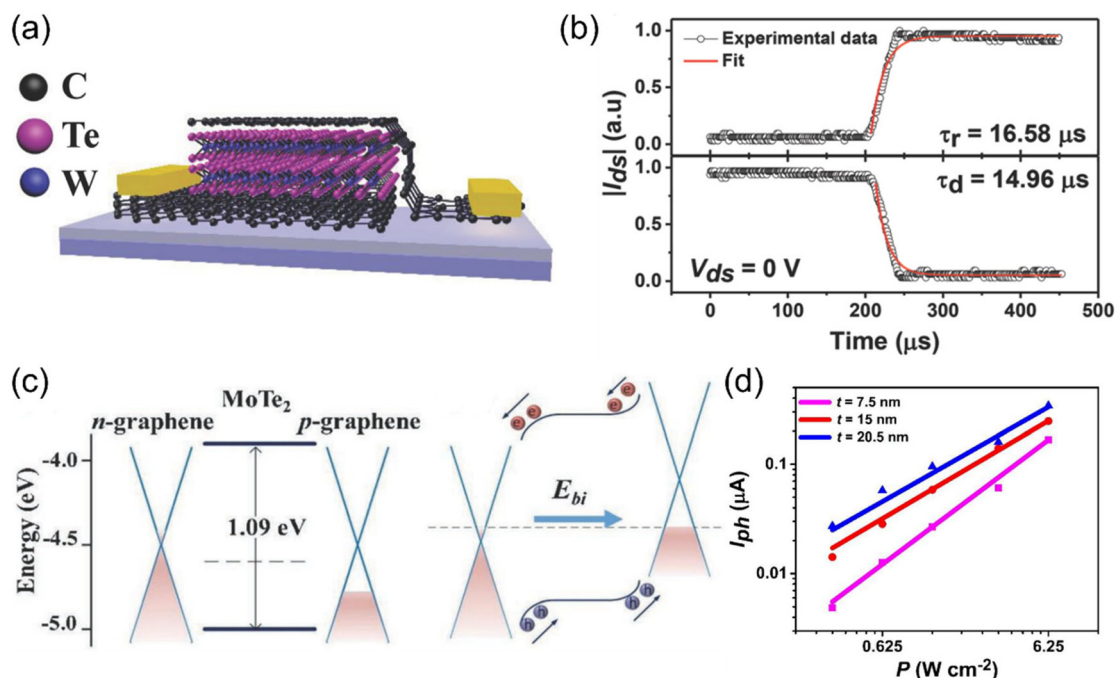


Fig. 7 (a) Schematic diagram of a vertical asymmetric Gr/MoTe₂/Gr self-driven photodetector. (b) Temporal photoresponse. The red lines are the fitting curves. (c) Energy band diagrams of isolated (left) and contacted (right) n-type exfoliated graphene, p-type MoTe₂, and p-type CVD-grown graphene. (d) Photocurrent as a function of light power intensity of Gr/MoTe₂/Gr photodetectors with different MoTe₂ channel thicknesses. The source–drain bias for (b) and (d) is zero. Reproduced from ref. 165 with permission from John Wiley and Sons, Inc, copyright 2019.

precision photolithography/electron-beam lithography technology. This significantly increases the complexity and technical difficulty of the preparation process, leading to a substantial decrease in practical device yield due to misalignment, thus restricting the feasibility in large-scale applications to some extent. In view of these insurmountable deficiencies, there is still an urgent need to explore and develop new device architectures to simplify the preparation process of self-driven 2DLM photodetectors, improving the yield and practicality.

To this end, in 2018, Zhou *et al.* developed an MSM-type self-driven WSe₂ photodetector with asymmetric metal contact geometries (Fig. 8a).⁷⁶ Specially, compared to the common MSM-type devices with identical contact length at two ends (Fig. 8b), the contact lengths of the asymmetric device in the study are controlled to be discrepant (Fig. 8a). As shown in Fig. 8c, under illumination, the asymmetric WSe₂ device demonstrates distinct photovoltaic characteristics, as evidenced by the I - V curves obviously bypassing the origin. In principle, the bias-free light-sensing characteristics can be explained as follows. Upon contact, a built-in electric field can be formed at the heterointerface of electrode and channel, on account of the different work functions of the two components. Under light illumination, photocurrent is generated under the drive of the built-in electric field. Of note, the photocurrent density of the two electrode/channel interface is identical since the interfacial potential is theoretically determined by the difference in the work functions of electrode and channel, while the directions are opposite due to the back-to-

back configuration. As for the symmetric configuration (Fig. 8b), the photocurrent flows generated from the two heterogeneous interfaces are exactly compensated, and thus no photoresponse can be observed. As for the geometrically asymmetric configuration, the heterointerface with longer contact length dominates, and net photocurrent flow (I_{SC}) can thus be generated according to

$$I_{SC} = J_{SC}(W_R - W_L)t, \quad (4)$$

where J_{SC} is the photocurrent density, W_R is the contact length of the right electrode/channel junction, W_L is the contact length of the left electrode/channel junction, and t is the channel thickness. As shown in Fig. 8d and e, the photovoltage and photocurrent increase with increasing light power, on account of the increased number of photocarriers. By contrast, the responsivity exhibits an opposite trend. Most importantly, the photoresponse can be flexibly modulated by the asymmetric level. As a rule, as the contact length difference increases, the photovoltage monotonously increases (Fig. 8f). On the whole, this research underscores a valuable approach for implementing MSM-type self-driven photodetectors in a straightforward and cost-efficient manner by breaking the geometrical symmetry. Following this success, self-driven photodetectors based on various 2D materials, such as In₂S₃,¹⁶⁶ InSe¹⁶⁷ and ReSe₂,¹⁶⁸ have been realized by using this strategy, demonstrating its broad universality.

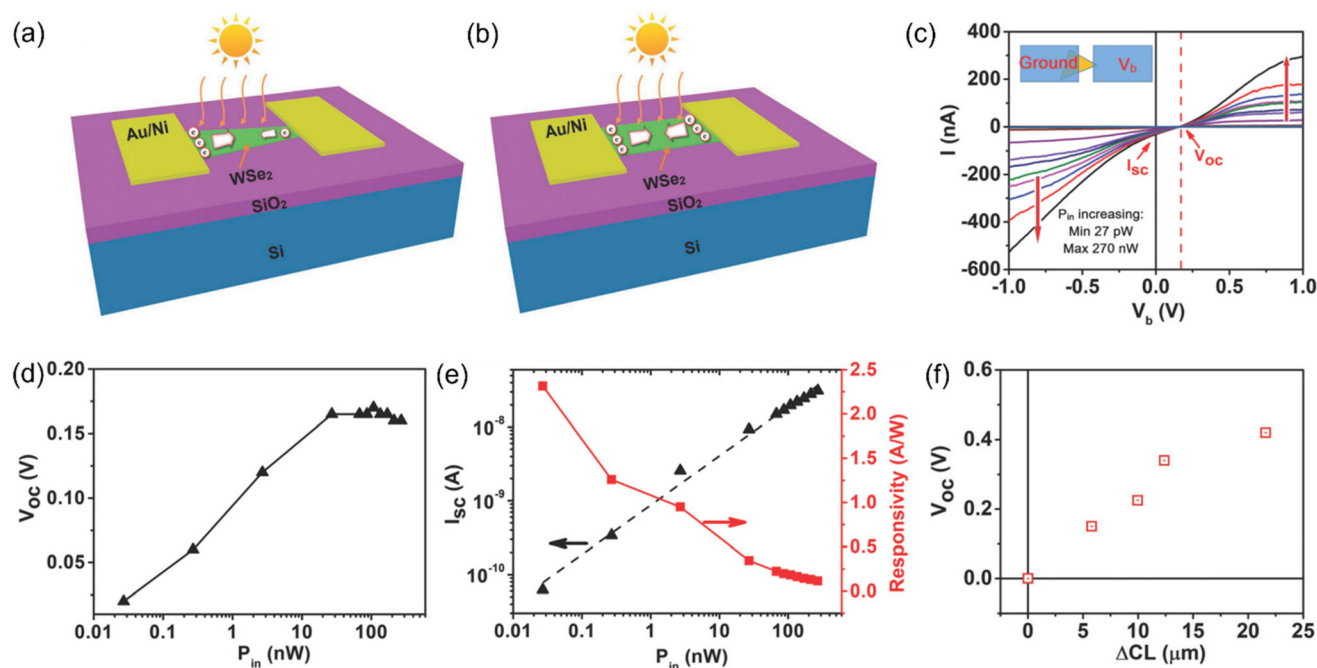


Fig. 8 Schematic diagrams of the MSM-type photodetectors with (a) asymmetric contact geometry and (b) symmetric contact geometry. (c) I - V characteristic curves of the asymmetric device under illuminations with various light powers. The inset shows the measurement setup. (d) The open-circuit voltage (V_{OC}) as a function of light power. (e) Light power dependent short-circuit current (I_{SC} , black dots and line) and the corresponding zero-biased responsivity (red dots and line). (f) Contact length difference (ΔCL) dependent V_{OC} . Reproduced from ref. 76 with permission from John Wiley and Sons, Inc, copyright 2018.

6.3 Asymmetric contacts with different contact areas

In addition to the contact length of electrode, the interfacial area shared between the electrode and the light-sensing channel constitutes another crucial ingredient influencing the transfer of photoexcited charge carriers across the heterointerface, thereby opening up an additional promising pathway towards realizing self-driven optoelectronic sensing through technologically streamlined two-terminal architectures.

As a demonstration, in 2023, Wang *et al.* transferred a rectangular Te nanosheet onto prepatterned Cr/Au electrodes with identical contact length but different contact areas at the two ends and demonstrated that the asymmetric device exhibited self-driven optoelectronic response.¹⁶⁹ The structure of the device is schematically shown in Fig. 9a, where the contact area between the drain electrode and the Te channel is significantly smaller than that between the source electrode and the Te channel. As shown in Fig. 9b, under illumination, the I_{ph} - V_d curves of the device evidently bypass the origin, which indicates a pronounced photovoltaic effect, providing a firm guarantee for self-driven optoelectronic sensing. Under a source-drain voltage of 0 V and periodic illumination, the device exhibits stable and reproducible on/off switching characteristics (Fig. 9c). By virtue of the extremely low dark current, the on/off ratio (I_{ph}/I_{dark}) under a self-driven mode is quite competitive, which is nearly three orders of magnitude higher than that under a biased operating mode (Fig. 9d). In addition, as the

incident light power increases, the on/off ratio monotonically increases, due to the increased number of photoexcited carriers. Taking advantage of the suppressed shot noise, the detectivity under a self-driven mode reaches $\sim 10^{12}$ Jones, which is 1–2 orders of magnitude higher than that under biased conditions (Fig. 9e). In addition, the response/recovery time without bias is as short as 720 μ s/1.16 ms (Fig. 9f). Fig. 9g shows the photocurrent mapping image. Evidently, the photocurrent is predominantly generated at the electrode/channel interfaces, and the directions of the photocurrent at the two ends are opposite. Of note, the photocurrent value in vicinity of the drain electrode is slightly higher than that in vicinity of the source electrode. Therefore, the physical mechanism of the self-driven characteristics of this asymmetric device can be explained as follows. Due to the lower Fermi level of Au as compared to Te, a built-in electric field pointing from Te towards Au will be formed after metallization. Under illumination, photoexcited electron-hole pairs are generated. Under the drive of the built-in electric field, photoelectrons near the source and drain electrodes drift towards the center of the Te channel. Due to the smaller contact area of the drain electrode as compared to the source electrode, there are fewer photoholes at the interface of the drain electrode, resulting in weaker attraction to photoelectrons and more efficient electron drift (Fig. 9h). Due to the discrepancy of electron drift at the two ends, zero-bias photoresponse is triggered. Overall, this study exemplifies a simple protocol free from complex micro-

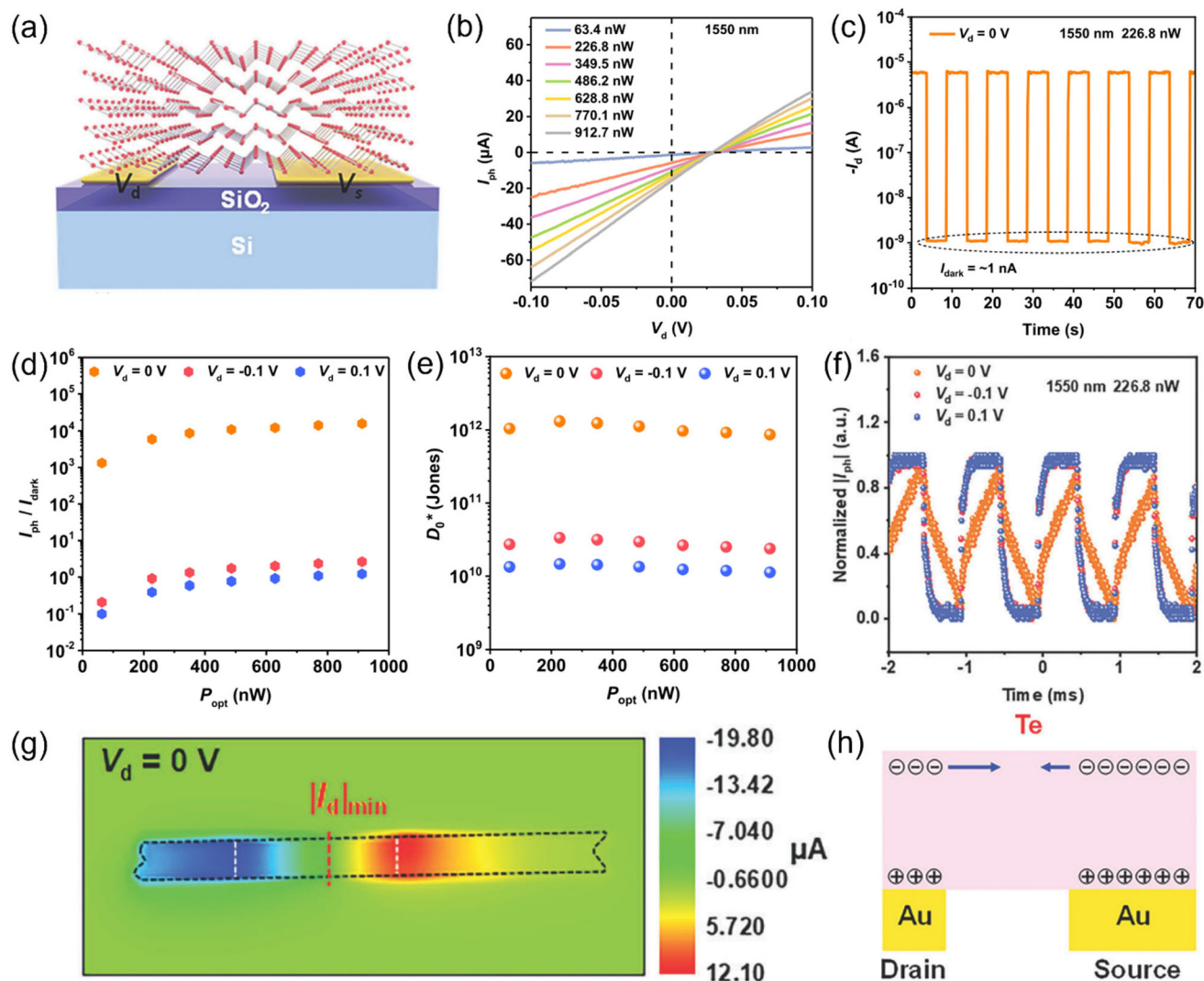


Fig. 9 (a) Schematic diagram of a self-driven Te photodetector with electrodes of different contact areas. (b) $I_{ph}-V_d$ curves under 1550 nm illuminations with various light intensities. (c) Photoswitching curve under periodic 1550 nm illumination. (d) I_{ph}/I_{dark} and (e) detectivity as a function of light power under various biased conditions. (f) Temporal photoresponse. (g) Photocurrent mapping image without external bias. (h) Schematic illustration of the working mechanism for self-driven photoresponse. Reproduced from ref. 169 with permission from John Wiley and Sons, Inc, copyright 2023.

processing techniques for the implementation of self-driven 2DLM photodetectors *via* geometrical nonequivalence of source and drain contacts while using the same metal for electrodes. Most recently, Ghanbari *et al.* have prepared a self-driven MoSe₂ photodetector based on an analogous scheme and achieved a remarkable responsivity of 4.38×10^7 V W⁻¹.¹⁷⁰

7. Orthogonal electrodes

In addition to the material and geometry of electrodes, the modulation of the relative locations of electrodes is also an effective approach to reduce the overall symmetry of the device, providing a distinctive pathway for the implementation of self-driven photodetection. As a demonstration, in 2023,

Semkin *et al.* fabricated a zero-bias photodetector with polarization discrimination capability by patterning metal contacts onto the orthogonal sides of a rectangular PdSe₂ channel (Fig. 10a).¹⁷¹ As shown in Fig. 10b and c, the device exhibits pronounced polarization-dependent self-driven photoresponse. Specially, when the polarization of incident light is perpendicular to the source electrode, photocurrent predominantly generates at the source electrode/channel interface (left panel, Fig. 10b). By contrast, when the polarization of incident light is perpendicular to the drain electrode, photocurrent predominantly generates at the drain electrode/channel interface (right panel, Fig. 10b). To have an in-depth insight, systematic FDTD simulations have been performed. As shown in Fig. 10d, it is unveiled that the light field intensity is greatly enhanced at the electrode/channel interface when the polarization direc-

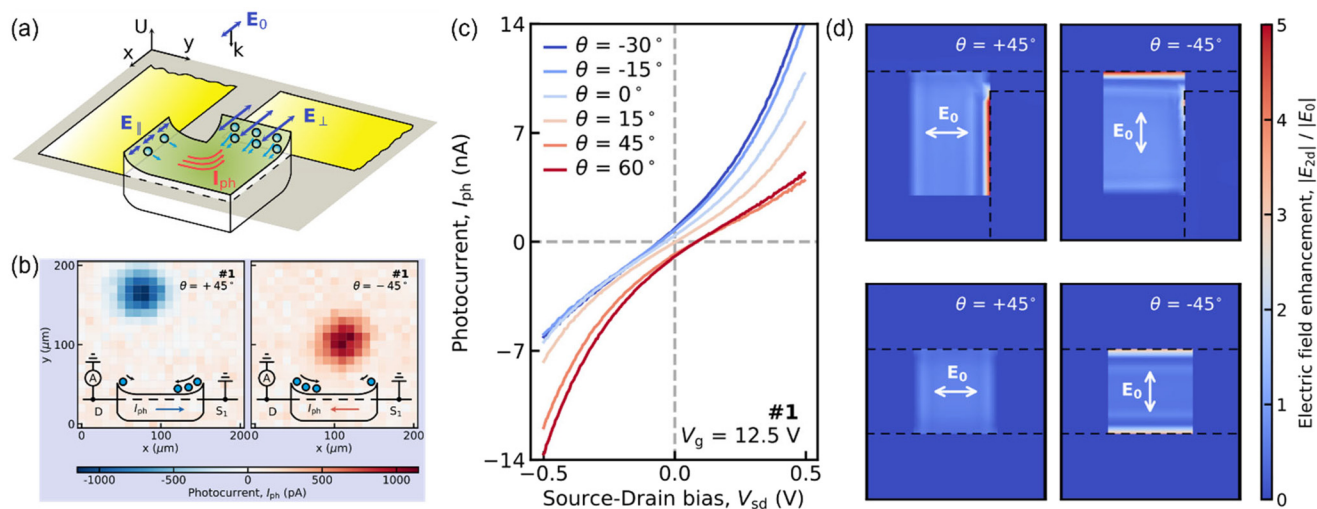


Fig. 10 (a) Schematic illustration of a self-driven PdSe₂ photodetector with orthogonal metal contacts (corner-type device). (b) Photocurrent mapping images upon linearly polarized illuminations with polarization angles of 45° (perpendicular to the source electrode/channel interface) and –45° (perpendicular to the drain electrode/channel interface). (c) Photocurrent as a function of source–drain bias upon light illuminations with various polarization directions. (d) Electric field enhancement of corner-type (top panel) and slit-type (bottom panel) devices. Reproduced from ref. 171 with permission from American Chemical Society, copyright 2024.

tion of incident light is perpendicular to the interface, which is associated with the lightning-rod effect.¹⁷² As for the proposed corner-type device, when one electrode is perpendicular to the polarization direction, the other one becomes parallel, which is thus selectively shadowed. As a consequence, the photocurrent flows generated at the two electrode/channel interfaces cannot be fully compensated, thus giving rise to self-driven photoresponse (Fig. 10a). On the whole, this study establishes a new scheme for realizing bias-free photodetection. Of note, since this strategy is independent of a particular material property, it is generic and can be theoretically applied to arbitrary 2DLMs in the upcoming future.

8. Asymmetric electrodes with discrepant thermal conductivities

Long-wave infrared (LWIR) photodetection has played a crucial role in remote sensing, infrared imaging, and other areas. Nonetheless, traditional 2DLM based LWIR photodetectors have been confronted with a series of dilemmas. In general, the photoconductive-type devices built of narrow-bandgap semiconductors/semimetals have suffered from large dark current, low signal-to-noise ratio, and high energy dissipation due to the numerous thermally excited free carriers. As a consequence, cryogenic cooling devices are generally required during operation to be competent for the standards of practical applications, which seriously increase device volume and operating cost. The photovoltaic devices, mostly built of p–n junctions or Schottky junctions, can address the above-mentioned predicaments by operating in a self-driven mode, where no source–drain bias is applied. However, there also exist some tricky challenges for this category of devices. For

example, since the density of free carriers of narrow-bandgap semiconductors/semimetals is relatively high, the depletion layer therein of related heterojunctions is quite narrow. In addition, it is more challenging to form a staggered type-II band alignment, a favorable configuration to drive the segregation of photocarriers, due to the small interval of the conduction band minimum and valence band maximum of small-bandgap semiconductors. As a consequence, photoexcited electron–hole pairs easily recombine without being spatially separated, thus leading to poor photosensitivity. In addition, Fermi-level-pinning free heterointerface is required, which has been a long-standing bottleneck. Therefore, self-driven 2DLM LWIR photodetectors with new working mechanism have been the subject of intensive research.

Photothermoelectric (PTE) detector is a distinctive type of device based on the Seebeck effect.¹⁷³ Basically, the PTE effect is consisted of two pivotal processes: photothermal conversion and thermoelectric conversion. In the photothermal conversion process, carriers absorb energy from photons, and then transfer the energy to the crystal lattice (*i.e.* thermal energy) through electron–phonon interactions. In the thermoelectric conversion process, the carriers are driven from the hot side to the cold side by the temperature gradient, leading to an electric potential difference (V) described by

$$V = \left| \int S(x) \nabla T(x) dx \right| \quad (5)$$

where $S(x)$ is the Seebeck coefficient, and $\nabla T(x)$ is the temperature gradient induced by the photon absorption. Accordingly, the generation of photovoltage can be achieved by inducing a temperature gradient across the photosensitive channel, which has been commonly realized by a focused light beam.^{174–177} However, in practical usage, global illumination is a more

common scenario, especially for the miniaturized 2DLM photodetectors. Accordingly, novel device structures should be investigated to induce temperature gradient across the 2DLM channel upon global illumination.

As an example, Dai *et al.* developed a PTE-type photodetector based on a semimetal PdSe₂ nanoflake and successfully achieved self-driven LWIR photodetection through integrating asymmetric electrodes.¹⁷⁸ The device structure of the MSM-type PdSe₂ photodetector is schematically illustrated in Fig. 11a, where multilayer graphene (MLG) and Au electrodes are utilized as the electrically asymmetric contacts. As shown in Fig. 11b–d, under zero source–drain bias and room temperature, the MLG/PdSe₂/Au device exhibits pronounced photoresponse to periodic 4.6, 7.4, and 10.5 μm global illuminations. The on/off switching is prompt, distinct, and well reversible, demonstrating great reliability. Of note, photovoltage mapping measurement has determined that the photoresponse is the most pronounced in vicinity of the electrode/

channel interface (Fig. 11e). In addition, the photovoltage in vicinity of the Au electrode is evidently higher than that in vicinity of the MLG electrode, suggesting that the PTE effect of the former is stronger than that of the latter. In consideration of this, the self-driven photoresponse characteristics can be associated with the asymmetric PTE effects. Upon global illumination, the PdSe₂ channel absorbs the photons and the temperature increases through photothermal conversion. Since the thermal conductivity of multilayer graphene ($\approx 3000 \text{ W m}^{-1} \text{ K}^{-1}$) is substantially higher than that of Au ($\approx 180 \text{ W m}^{-1} \text{ K}^{-1}$), the thermal dissipation of the former is much more efficient than that of the latter, thereby resulting in a lateral temperature gradient along the PdSe₂ channel (Fig. 11f). As a consequence, photovoltage is generated due to the Seebeck effect. Moreover, by virtue of the anisotropic light absorbance enabled by the in-plane anisotropic crystal structure of 2D PdSe₂, the self-driven MLG/PdSe₂/Au photodetector has manifested pronounced polarized photoresponse, portending the

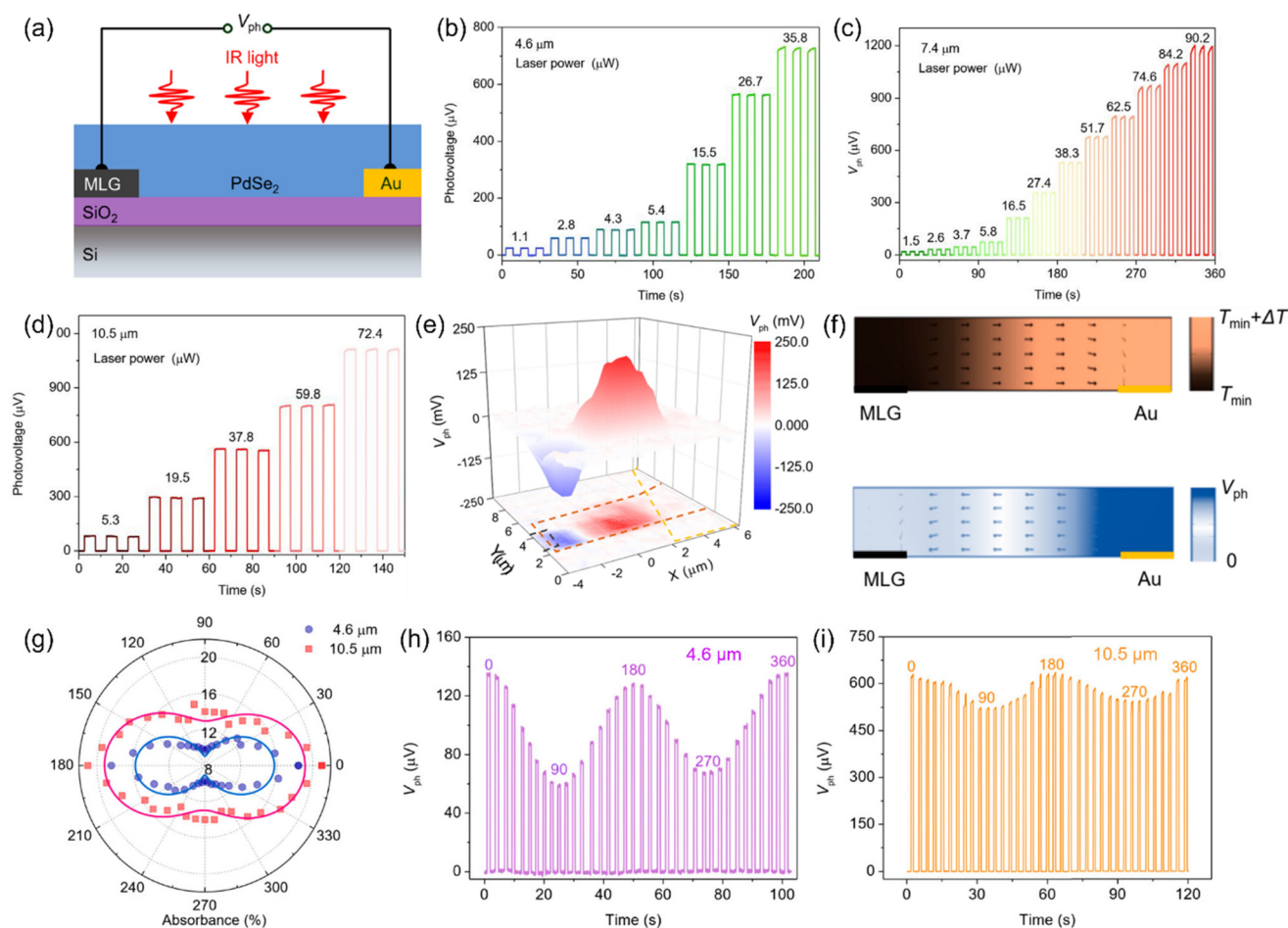


Fig. 11 (a) Schematic diagram of an MSM-type PdSe₂ photodetector with multilayer graphene (MLG) and Au as asymmetric electrodes. (b–d) Photoswitching curves under (b) 4.6 μm , (c) 7.4 μm , and (d) 10.5 μm periodic illuminations with different laser powers at zero bias. (e) 3D mapping plot of photovoltage (V_{ph}) under 532 nm illumination. (f) Schematic diagrams of temperature (top panel) and photovoltage (bottom panel) distribution of the PdSe₂ photodetector under global light illumination. The gradients of temperature and electric field are indicated by the black and blue arrows, respectively. (g) Polar plots of light absorbance as a function of the polarization angle of PdSe₂ for 4.6 μm (blue line and circles) and 10.5 μm (red line and squares) incident light. (h–i) Polarized photovoltage response under (h) 4.6 μm and (i) 10.5 μm illuminations. Reproduced from ref. 178 with permission from American Chemical Society, copyright 2022.

prodigious potential for photodetection in complex environments and multifunctional optoelectronic applications (Fig. 11g–i). In general, this study unveils a facile and generic scheme, circumventing the necessity to realize Fermi-level-pinning-free heterointerface, toward self-driven LWIR 2DLM photodetectors by harnessing the Seebeck effect. It is noteworthy that despite the considerable disparity in the thermal conductivities of the two electrodes, Au also stands out as an exceptional heat sink, which hinders the establishment of a substantial temperature gradient between the two ends of the channel. Looking ahead, the conductors with low thermal conductivity (e.g., $\text{W}_x\text{V}_{1-x}\text{O}_2$ ¹⁷⁹) could be leveraged to substitute conventional metals, thereby amplifying the thermal conductivity contrast between the electrodes and consequently enhancing the photoresponse of the device even further.

In addition to integrating asymmetric electrodes with different thermal conductivities to trigger a temperature difference, introducing a Seebeck coefficient gradient, through various ingredients such as contact doping effect enabled by electrodes with different work functions, is also a potential approach to achieve self-driven optoelectronic sensing.¹⁸⁰ As a prototypical example, in 2020, Guo *et al.* demonstrated self-driven photodetection by depositing Cr/Au and Ti/Au electrodes at the two ends of a black phosphorus (BP) nanosheet (Fig. 12a).¹⁸¹ Fig. 12b summarizes the photocurrent *vs.* source-drain voltage ($I_{\text{ph}}-V$) curves of the black phosphorus photo-

detectors with symmetric and asymmetric electrodes under 0.12 THz illumination. Evidently, under a source-drain bias of 0 V, the symmetric Cr/Au-BP-Cr/Au device exhibits negligible photoresponse, whereas the asymmetric Cr/Au-BP-Ti/Au device boasts distinct photocurrent, indicating that the asymmetric electrode configuration causes the self-driven photoelectric response. Under the irradiation of terahertz waves of different frequencies, the asymmetric black phosphorus device exhibits pronounced on/off switching characteristics (Fig. 12c). Specifically, upon 0.12 THz illumination, the rise time and decay time are as short as 0.8 and 1.5 μs , respectively (Fig. 12d). Overall, the self-driven optoelectronic characteristics of the Cr/Au-BP-Ti/Au device can be attributed to the symmetry breaking of Seebeck coefficient of the black phosphorus channel caused by different electrodes. As a rule, the Seebeck coefficient (S) of a material is closely related to the Fermi level (E_{F}) and can be described by the following formula¹⁸²

$$S = \frac{\pi^2 k_{\text{B}}^2 T}{6eE_{\text{F}}}, \quad (6)$$

where π is the circular constant, k_{B} is the Boltzmann constant, T is temperature, and e is elementary charge. In principle, the asymmetric electrodes can cause shift of the Fermi levels of adjacent black phosphorus channels through charge transfer, thus resulting in difference in the Seebeck coefficient at the

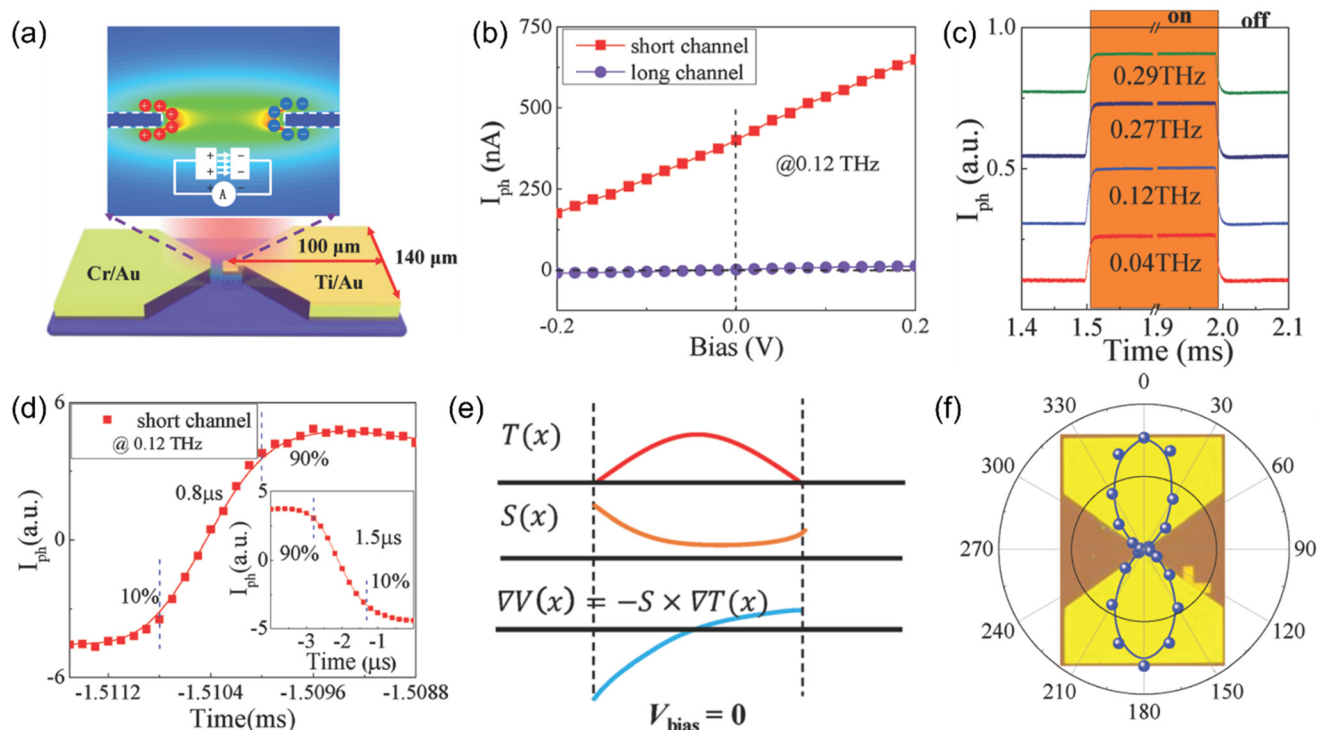


Fig. 12 (a) Schematic illustration of a black phosphorus photodetector with asymmetric Cr/Au and Ti/Au contacts. (b) $I_{\text{ph}}-V$ curves of black phosphorus photodetectors with symmetric (blue line) and asymmetric (red line) contacts. (c) Photoswitching curves upon terahertz illuminations with different frequencies. (d) Temporal photoresponse. (e) Schematic illustration of the spatial distribution of temperature, Seebeck coefficient, and electric potential. (f) Polar plot of the photocurrent as a function of polarization angle. Reproduced from ref. 181 with permission from John Wiley and Sons, Inc, copyright 2020.

two ends of the channel. Therefore, the underlying mechanism of the self-driven optoelectronic response of the device can be explained as follows. Under irradiation, due to the heat dissipation effects of the electrodes, the temperature in the middle of the BP channel is higher than those at both sides (Fig. 12e). As a consequence, PTE effect is generated due to Seebeck effect according to eqn (4). Due to the different Seebeck coefficients at both ends of the BP photosensitive channel, self-driven photoelectric response is induced. Furthermore, since that the electrodes with deep subwavelength gap can markedly increase the electric field strength of parallelly polarized electromagnetic waves, the Cr/Au-BP-Ti/Au device exhibits distinct polarized photosensitivity (Fig. 12f), portending enormous potential in multi-dimensional optical information acquisition. Due to the fact that the self-driven light-sensing characteristics originate from asymmetric electrodes rather than channel, this strategy theoretically is applicable to various 2DLMs. Actually, in another study, Cai *et al.* also realized self-driven photodetection by integrating Cr and Au electrodes at the two ends of a graphene channel,¹⁸³ underscoring the broad applicability of Seebeck coefficient engineering.

In addition to electrodes, dielectric layers can also have a significant effect on the Seebeck coefficient of 2DLMs, providing another pathway for the implementation of self-driven 2DLM photodetectors. For a prototypical example, in 2023, Razeghi *et al.* achieved a single material MoS₂ thermoelectric junction by substrate engineering by locating a MoS₂ nanosheet onto a holey substrate.¹⁸⁴ It turns out that the self-driven photoresponse can be generated in a supported MoS₂/suspended MoS₂ homojunction. This can be explained by the difference in Seebeck coefficient on account of the variation in the dielectric environment, which alters both the electron-phonon scattering mechanism and the Fermi level of MoS₂.

9. Ferroelectricity-enabled bulk photovoltaic effect

As previously elaborated, the upper limit of photovoltaic output achievable by traditional p-n junctions is dictated by the renowned Shockley–Queisser theory, posing a formidable barrier to advancing the device properties. Consequently, it becomes imperative to embark on exploring novel strategies aimed at attaining self-driven operation. Ferroelectric materials are unique substances characterized by spontaneous polarization, which can be readily adjusted by an external electric field. Typically, these materials exhibit low-symmetry crystalline structures in which the centers of positive and negative charges fail to overlap. This inherent structural asymmetry engenders the emergence of an internal electric field, oriented in opposition to the direction of the ferroelectric polarization. Since the ferroelectric polarization is nonvolatile, the resulting internal electric field can be exploited as the impetus to drive the spacial separation of photocarriers, a process recognized as the BPVE. In this consideration, ferroelectricity opens up new possibilities for realizing bias-free 2DLM photodetectors.

As proof of the potency, in 2024, Li *et al.* constructed a self-driven 2DLM photodetector by exploiting the ferroelectricity-enabled BPVE, in which a Sn₂P₂S₆ nanosheet is sandwiched vertically between two thin graphene layers, as schematically shown in Fig. 13a and b.¹⁸⁵ As presented in Fig. 13c, the poled Sn₂P₂S₆ device exhibits stable photoresponse to periodic light simulations. In addition, the direction of the photocurrent can be flexibly switched by inverting the ferroelectric polarization direction, confirming the ferroelectricity origin of the self-driven optoelectronic characteristics. Essentially, the ferroelectric polarization inherent in the vertically poled Sn₂P₂S₆ nanosheet establishes a robust built-in electric field extending along the out-of-plane axis. Upon illumination, photogenerated electrons and holes within the Sn₂P₂S₆ matrix are efficiently segregated along opposite trajectories, thereby spontaneously triggering a photoresponse without the need for external biasing. Furthermore, taking advantage of the polarization-sensitive light absorption (Fig. 13d), the Sn₂P₂S₆ photodetector exhibits distinct polarized photosensitivity (Fig. 13e). Specifically, by using the Sn₂P₂S₆ device as the sensing component, polarization-sensitive imaging has been realized (Fig. 13f). Broadly speaking, this study presents a straightforward pathway that sidesteps the intricate methodologies typically required for crafting high-quality p-n junctions, Schottky barriers, and homojunctions, thereby facilitating the realization of self-driven 2DLM photodetectors in a more accessible manner. Thus far, a series of ferroelectric 2DLMs, including CuInP₂S₆,¹⁸⁶ SnS,^{187,188} ReS₂,¹⁸⁹ SnP₂Se₆,¹⁹⁰ and α -In₂Se₃,^{191,192} have been successfully harnessed for self-driven photodetection by exploiting the similar mechanism.

As aforementioned, self-driven 2DLM photodetectors, which harness BPVE as the underlying operating mechanism, commonly necessitate the employment of channel materials that possess intrinsic ferroelectricity. Nevertheless, a substantial portion of 2DLMs, notably the widely explored transition metal dichalcogenides, possess relatively high crystalline symmetry, which inherently limits the manifestation of ferroelectric properties. Fortunately, by meticulously manipulating the growth direction, it is feasible to degrade the overall symmetry of 2DLMs, thereby unleashing the potential for ferroelectric behavior as well as BPVE devices.

In their most recent endeavor, Xue and colleagues successfully accomplished the oriented growth of WS₂ via an atomic manufacturing strategy (Fig. 14a), yielding diverse nanostructures such as zigzag, armchair, and chiral ribbons.¹⁹³ Remarkably, the armchair-structured WS₂ photodetector demonstrates a pronounced self-driven photoelectric response, evidenced by the *I*-*V* curves obviously detouring the origin upon illumination (Fig. 14b). Furthermore, the device displays a noteworthy polarization photosensitivity, with the photocurrent manifesting a periodic variation as the polarization angle varies from 0 to 360° (Fig. 14c). Notably, the photocurrent attains the maximum value when the polarization direction aligns parallel to the growth orientation, whereas it diminishes to the nadir when the polarization direction is perpendicular to the growth axis. Further investigations have

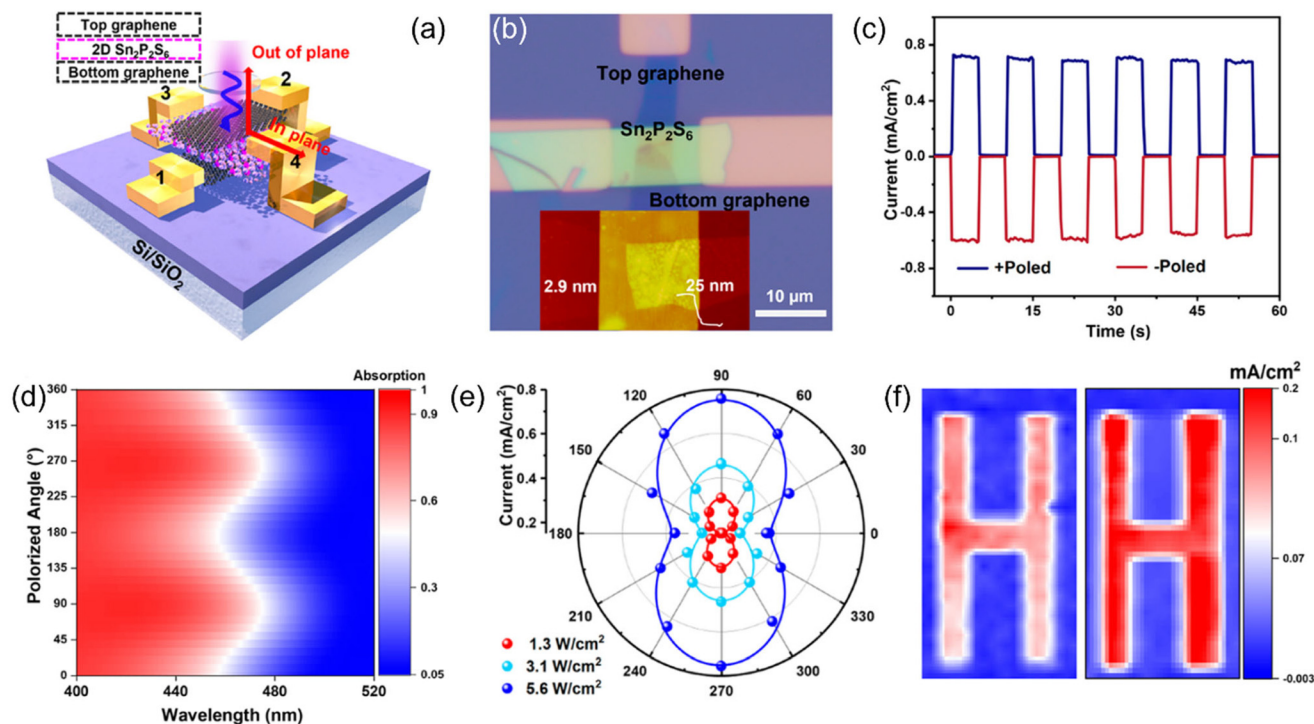


Fig. 13 (a) Schematic diagram of a vertical Gr/Sn₂P₂S₆/Gr photodetector. (b) A typical optical microscopy image. The inset shows the AFM topography image and the height profiles. (c) Photoswitching curves at zero bias under different poled states. (d) Contour mapping of the polarized light absorption of Sn₂P₂S₆. (e) Polar plots of the photoresponse as a function of polarization angle under illuminations with various light powers. (f) The imaging results with the polarization angles of 0° (left panel) and 90° (right panel). Reproduced from ref. 185 with permission from American Chemical Society, copyright 2024.

unveiled that the photoresponse of the armchair WS₂ photodetectors is much higher than those of the chiral and zigzag devices (Fig. 14d). This compelling finding suggests that the self-driven photoresponse characteristics of the armchair devices stem from the ferroelectricity induced by the degraded symmetry. When WS₂ grows along the armchair direction, the equivalent negative charge center formed by S atoms fail to overlap with the equivalent positive charge center formed by W atoms. Therefore, an internal electric field is generated (Fig. 14e). After photoexcitation of electron-hole pairs, the electrons and holes migrate in the opposite directions under the drive of the internal electric field, resulting in the generation of photocurrent. In summary, this pioneering research introduces a novel perspective for the realization of self-driven photodetectors, which heralds the potential for extensive application to additional 2DLMs, including MoS₂, MoSe₂, and WSe₂, in the imminent future.

10. Self-driven photodetectors enabled by asymmetric optical antennas

As mentioned above, provided 2DLMs boast intrinsic ferroelectricity, the non-volatile polarization induced electric field can

spontaneously drive photocarriers and lead to the formation of bulk photovoltaic effect (BPVE), thereby enabling self-driven photodetection. However, in reality, most 2DLMs, such as graphene and MoS₂, have negligible intrinsic ferroelectricity. In this consideration, the strategy of ferroelectric engineering for self-driven photoelectric sensing suffers from relatively poor universality. In response to this predicament, it is thus necessary to explore an alternative strategy devoid of ferroelectricity to achieve self-driven photodetection by exploiting the bulk photovoltaic effect.

As an example, in 2020, Wei *et al.* demonstrated self-driven photoelectric detection by integrating non-centrosymmetric plasmonic optical antennas onto a graphene channel.¹⁹⁴ The structure of the device is schematically shown in Fig. 15a. A metasurface, which is consisted of periodic subwavelength and non-centrosymmetrical T-shaped Au patterns, has been prepared on a graphene channel by using standard electron beam lithography technique. Fig. 15b summarizes the output curves in darkness and under various illuminations. Specifically, compared to the *I-V* curve in the darkness, the curves under illuminations manifest a significant parallel shift, indicating that the non-centrosymmetric device possesses self-driven photoelectric response. In addition, under polarized light irradiation with polarization angles of 0 and 90 degrees, the skewing directions of the *I-V* curves are opposite, indicating that the device possess polarization sensitivity.

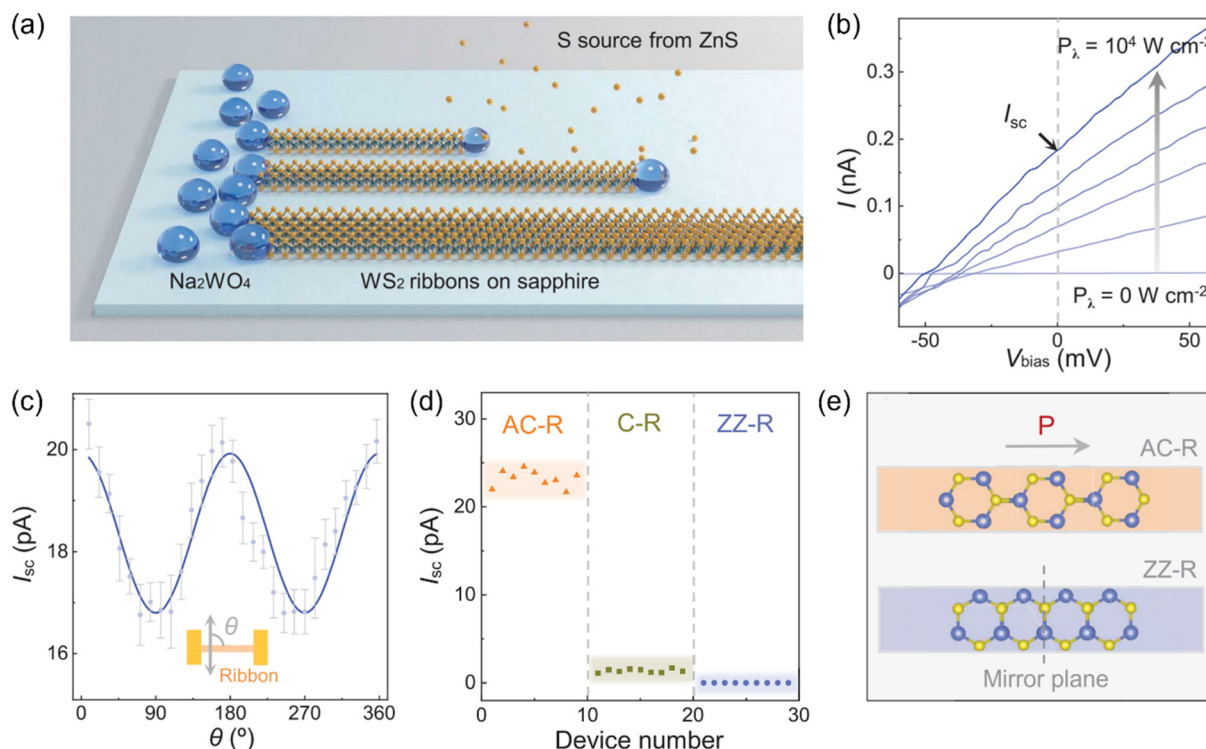


Fig. 14 (a) Schematic illustration for the oriented growth of WS₂. (b) *I*-*V* curves in darkness and under illuminations with various light power intensities. (c) Photocurrent as a function of polarization angle. The inset showcases the schematic definition of the polarization angle. (d) Photocurrent values of the armchair (AC-R), chiral (C-R), and zigzag (ZZ-R) WS₂ photodetectors. (e) Schematic comparison of the polarization states of armchair and zigzag WS₂. Reproduced from ref. 193 with permission from American Association for the Advancement of Science, copyright 2024.

Under periodic illumination, the device exhibits fast, stable, and reproducible photoresponse (Fig. 15c). Its response/recovery time is as short as $\approx 100/100$ μ s. Finite-different time-domain (FDTD) simulations determine that the self-driven photoresponse of the device is mainly due to the non-uniformity of the light field intensity, induced by the non-centrosymmetric T-shaped plasmonic optical antennas (Fig. 15d). As a consequence, directional net photocarriers are generated, leading to the generation of photocurrent. Similarly, in another following study, by modifying a graphene channel with double L-shaped nanoantennas, Xie *et al.* demonstrated high-performance long-wave infrared zero-biased photodetection with a high responsivity of 6.3 V W⁻¹ and a low noise-equivalent power of 1.6 nW Hz^{-1/2}.¹⁹⁵ Most recently, Koepfli and colleagues showcased a graphene-based photodetector that exhibited remarkable performance characteristics, including a high responsivity of 1.5 V W⁻¹ and an impressive operational frequency exceeding 400 GHz, achieved through the coupling of a "T"-shaped metamaterial perfect absorber architecture.¹⁹⁶ Overall, these studies have provided a new strategy for self-driven photodetection through the bulk photovoltaic effect unconstrained by specific family of materials.

In addition to integrating non-mirror-symmetric optical antennas within the 2DLM photosensitive channels, constructing electrodes with different light-harvesting capabilities is also an alternative protocol for achieving self-driven photode-

tection. As an example, most recently, Zhang *et al.* designed asymmetric bowtie split ring (BSR) and common rectangular electrodes on the two ends of a graphene channel (Fig. 16a), and achieved self-driven photodetection based on this non-centrosymmetrical device.¹⁹⁷ The *I*-*V* curves of the device under different conditions have been summarized in Fig. 16b. Apparently, compared to the *I*-*V* curve in darkness, the *I*-*V* curve under illumination shows a significant parallel shift away from the origin, indicating self-driven photoresponse characteristics. Specifically, the optimal responsivity reaches 138 V W⁻¹ (Fig. 16c), and the optimal noise equivalent power (NEP) is as low as 25 pW Hz^{-1/2} (Fig. 16d). The self-driven photoresponse of the asymmetric graphene device is mainly emanated from the localized light field enhancement effect enabled by the BSR electrode. As shown in Fig. 16e, profited from the inductor-capacitor and dipole resonances, the light field intensity on the BSR electrode side is significantly enhanced. Therefore, the corresponding light absorption is much stronger as compared to the other end. As a result, a temperature gradient is generated between the two ends, causing the self-driven photothermoelectric response *via* Seebeck effect (bottom panel of Fig. 16a). Most recently, Zhang *et al.* have achieved self-driven polarization-sensitive photodetection by integrating stripe- and triangle-shaped electrodes on the two ends of a graphene channel.¹⁹⁸ On the whole, the findings presented herein herald optical engineering as a promis-

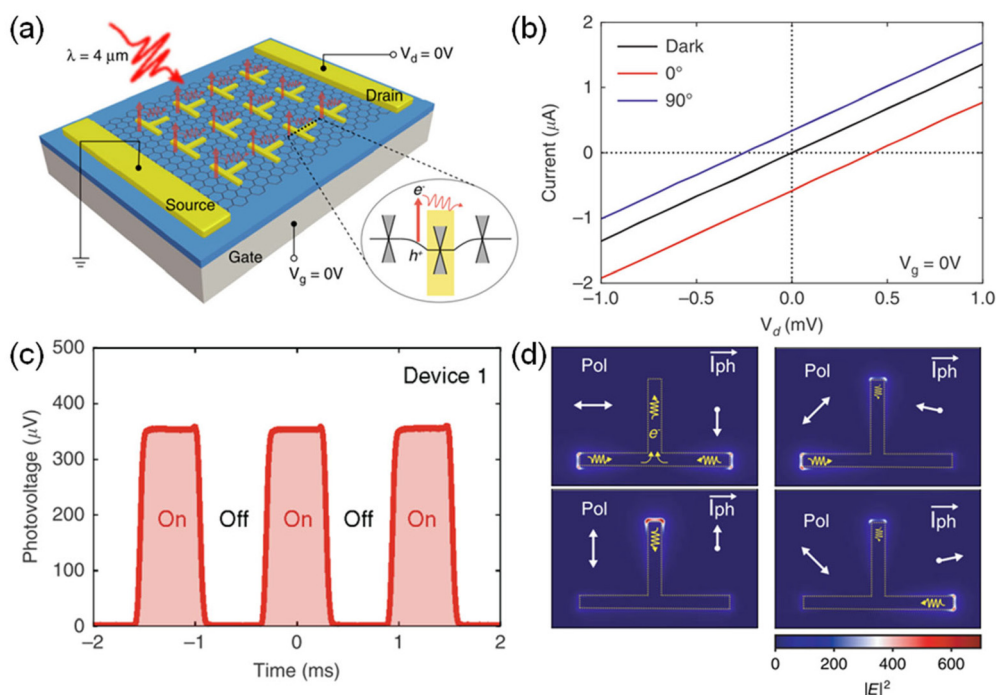


Fig. 15 (a) Schematic diagram of a self-driven graphene photodetector enabled by non-centrosymmetrical T-shaped plasmonic optical antennas. (b) I – V curves in darkness (dark line) and under light illuminations with various polarization directions (red & purple lines). Polarization angle: the angle between the direction of polarized light and the direction of source–drain bias. 0° : red line. 90° : purple line. (c) Photoswitching curve under periodic illumination at a source–drain bias of 0 V. (d) Simulated spatial distribution of the light field intensity upon light excitation with various polarization directions (indicated by the double-headed arrows). Yellow wave arrow: flow of photocarriers. White arrow: the resultant vectorial photocurrent. Reproduced from ref. 194 with permission from Nature Publishing Group, copyright 2020.

ing scheme to enable self-driven photodetection. As is universally acknowledged, the resonance characteristics of plasmonic optical structures are intricately tied to a multitude of ingredients encompassing shape, scale, and periodicity.¹⁹⁹ In the future, further refining optical antennas ought to be undertaken with the assist of numerical simulations, aiming to more precisely align them with the incident light and attain optimal light harvesting.

11. Self-driven photodetectors enabled by asymmetric carrier scattering

As previously noted, the realization of self-driven photodetectors inherently necessitates harnessing the built-in electric fields to segregate photogenerated electron–hole pairs. Nevertheless, the establishment of built-in electric field commonly entails the intricate design of heterojunctions or homo-junctions, which have been frequently plagued by serious interface challenges (contaminations, deformations, *etc.*). Most notably, the fabrication of these structures involves costly, complex, and labor-intensive processes, thereby impeding their widespread adoption. Consequently, researchers have been relentlessly pursuing innovative approaches to develop

interfacial built-in electric field-free self-driven 2DLM optoelectronic devices to overcome these limitations.

As a prototypical example, in 2020, Guo *et al.* achieved bias-free terahertz detection based on anisotropic carrier scattering of 2D PdTe_2 .²⁰⁰ Fig. 17a shows the structure diagram of the device. An MSM-type configuration is exploited, with a PdTe_2 nanosheet as the photosensitive channel and Cr/Au pads as the electrodes. As shown in Fig. 17b, the device exhibits stable response to terahertz light at a source–drain voltage of 0 V. By virtue of the excellent air-stability of PdTe_2 , the photoresponse remains largely invariable after one month of storage. In addition, the device exhibits a fast response rate, with a response/recovery time as short as $1/2.2 \mu\text{s}$ (Fig. 17c). As shown in Fig. 17d, the photocurrent is almost linearly related to the incident light power density, and the corresponding responsivity reaches $\approx 0.2\text{--}0.3 \text{ A W}^{-1}$. In general, the self-driven photoresponse of the device is mainly originated from asymmetric carrier scattering. As schematically shown in “i” of Fig. 17e, in darkness, the disordered movement of charge carriers causes scattered charge carriers in all directions to compensate each other, resulting in a negligible net current. Under polarized terahertz illumination, charge carriers move back and forth along the polarization direction, under the drive of the ac electric field. Since PdTe_2 boasts a C_{3v} symmetry point group, the carrier elastic scattering can be deemed as scattering by randomly distributed wedges with consistent orientation, and the

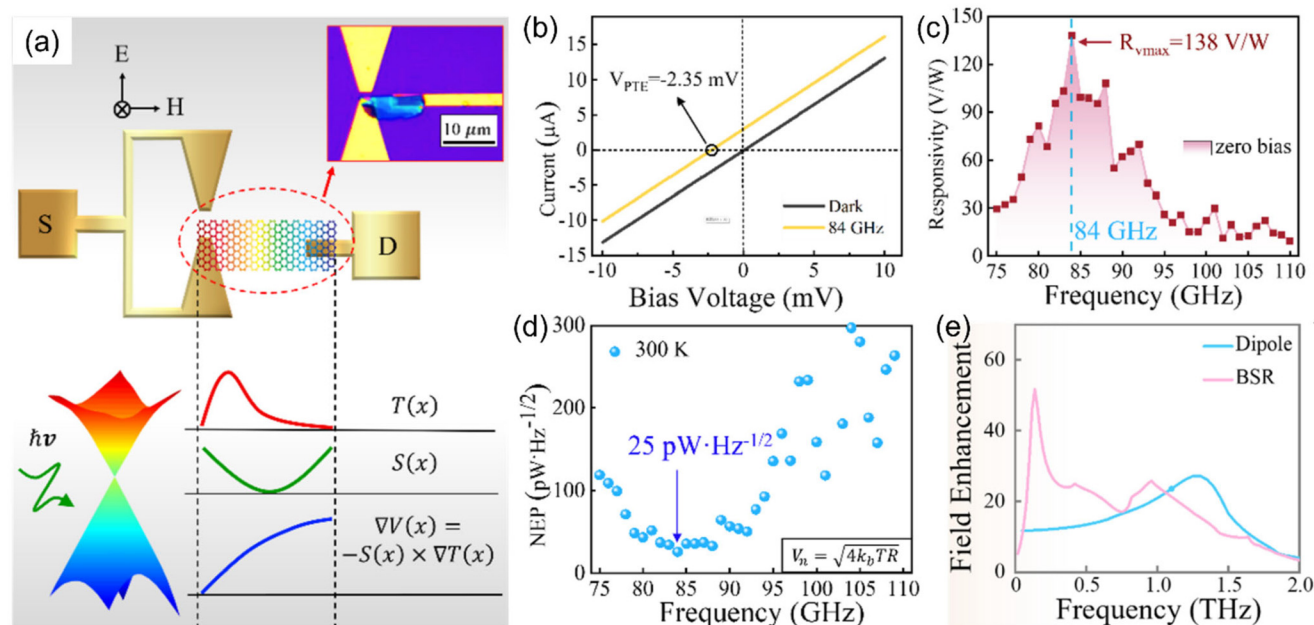


Fig. 16 (a) Schematic diagram of a self-driven graphene photodetector with asymmetric electrodes (top panel). The spatial distributions of temperature, Seebeck coefficient and electric potential have been summarized in the bottom panel. (b) I - V curves in darkness (dark line) and under 84 GHz light illumination (orange line). (c) Responsivity and (d) NEP as a function of the frequency of incident light. (e) Field enhancement as a function of frequency of dipole (blue line) and BSR (pink line) resonances. Reproduced from ref. 197 with permission from American Chemical Society, copyright 2022.

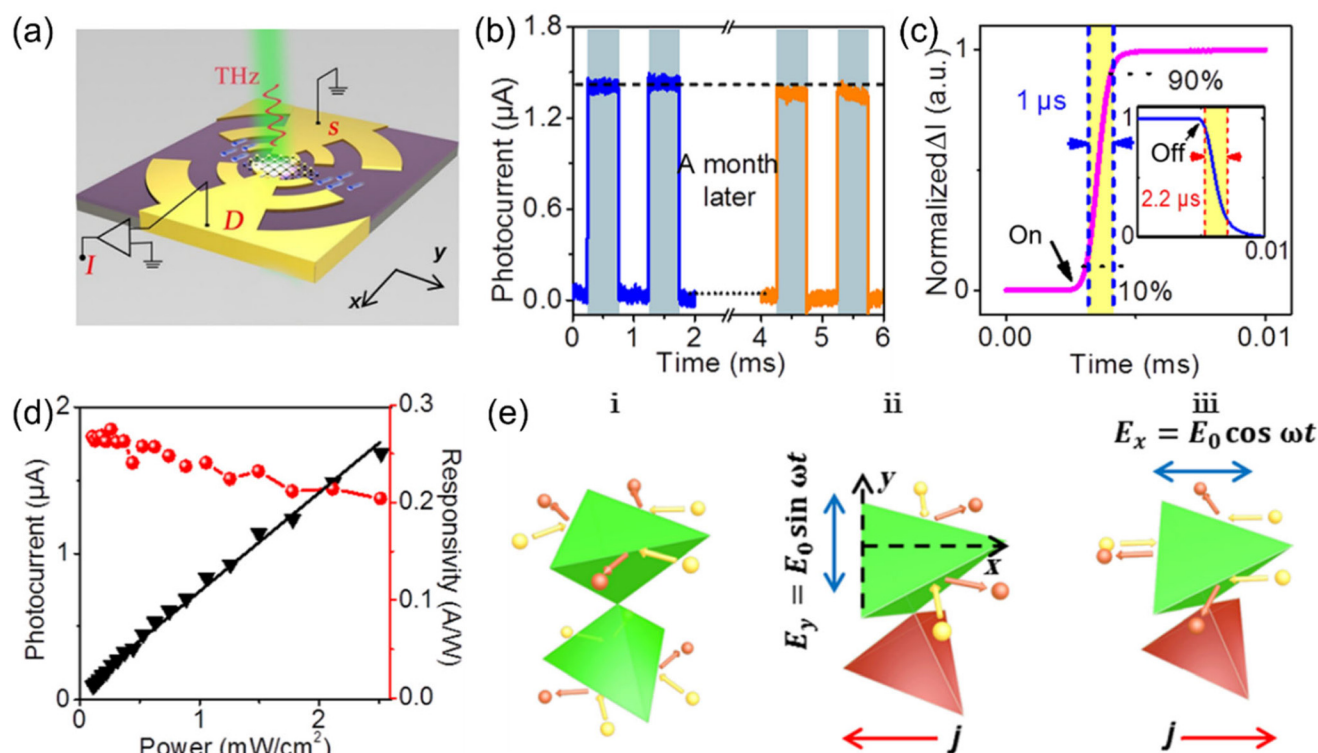


Fig. 17 (a) Schematic diagram of an MSM-type PdTe₂ terahertz detector. (b) Photoswitching curves under zero bias of a PdTe₂ device right after fabrication (blue line) and after storage in air for one month (orange line). (c) Temporal photoresponse. (d) Photocurrent (black line and triangles) and responsivity (red line and circles) as a function of light power density. (e) Schematic illustration of the working mechanism for the self-driven photoresponse. Reproduced from ref. 200 with permission from American Association for the Advancement of Science, copyright 2020.

scattering is highly anisotropic. Specifically, a polarization parallel to the base of the wedges yields the net electron flow perpendicular to the base, along the normal direction (ii, Fig. 17e). By contrast, a polarization perpendicular to the base of the wedges yields the net electron flow in an opposite direction (iii, Fig. 17e). As a consequence, self-driven photoresponse can be achieved, manifesting pronounced polarization sensitivity. Capitalizing on an analogous working mechanism, self-driven detectors have been implemented by using a series of 2DLMs, including Bi_2Se_3 ,²⁰¹ Sb_2Te_3 ,²⁰² and MnBi_2Te_4 ,²⁰³ substantiating the extensive applicability and versatility of this innovative strategy.

12. Conclusion & perspectives

A systematic overview on the new paradigms of self-driven photodetectors built of 2DLMs has been provided. Firstly, the

emerging strategies for implementing various self-driven 2DLM photodetectors have been epitomized, including strain modulation, thickness tailoring, structural engineering, asymmetric ferroelectric gating, asymmetric contacts (including work function, contact length, and contact area), ferroelectricity-enabled bulk photovoltaic effect, asymmetric optical antennas, *etc.*, with the focus on the fundamental physical principles of the self-driven characteristics. For clarity, Table 1 has summarized the pivotal figures-of-merit of the emerging self-driven 2DLM photodetectors. Following the above summarization, the ongoing challenges in this research field have been identified, and potential solutions addressing them have been proposed. On the whole, this review depicts a comprehensive blueprint for the implementation of self-driven, high-performance, and highly integrated 2DLM optoelectronic devices, which marks the advent of a new era of 2D optoelectronics. Nevertheless, despite the rapid development, the research regarding new paradigms of 2DLMs based self-driven

Table 1 Pivotal figures-of-merit of the burgeoning self-driven 2D material photodetectors

Device	R (A W^{-1})	EQE (%)	D^* (Jones)	$\tau_{\text{rise}}/\tau_{\text{decay}}$	Ref.
Thin-MoTe ₂ /thick-MoTe ₂	0.63	N. A. ^a	1.38×10^9	15 ms/17 ms	78
Thin-WSe ₂ /thick-WSe ₂	0.0112	2.6	4.4×10^{10}	0.18 ms/0.22 ms	107
Rolled WSe ₂ /flat WSe ₂	0.0346	N. A.	3.24×10^9	23.1 ms/16.8 ms	117
Ferroelectricity-enabled MoTe ₂ homojunction	1.5	3	N. A.	N. A.	120
Ferroelectricity-enabled MoTe ₂ homojunction	5	40	3×10^{12}	30 μs	121
Ferroelectricity-enabled MoS ₂ homojunction	0.015	N. A.	N. A.	20 μs	122
Ferroelectricity-enabled BP homojunction ^b	1.06	90.8	1.27×10^{11}	361 μs /362 μs	124
Ferroelectricity-enabled MoTe ₂ homojunction	0.22	41	N. A.	N. A.	125
Au/InSe/In	0.369	≈ 100	N. A.	23 ms/25 ms	131
Au/MoS ₂ /Ag	0.07564	15	1.27×10^{12}	8.85 ms/0.96 ms	133
Mo ₂ C/MoS ₂ /Au	10^{-4}	N. A.	N. A.	23 s/28 s	135
Cr/GeSe/Pd ^c	0.28	N. A.	4.1×10^9	69 ms/92 ms	147
MoO ₃ /MoSe ₂ /Ag	9.79×10^{-3}	2.3	6.23×10^9	5 ms/6 ms	137
1T'-MoTe ₂ /MoS ₂ /Cr	0.088	20	N. A.	N. A.	144
1T'-MoTe ₂ /InSe/Cr	0.048	N. A.	1.6×10^{12}	29 ms/29 ms	146
PtSe ₂ /WSe ₂ /Au	0.316	61.8	2.1×10^{11}	830 μs /950 μs	148
Cr/BP/WTe ₂	0.283	67	2.3×10^{11}	N. A.	150
Au/WSe ₂ /Gr	0.286	98	2.6×10^{12}	48 ms/45 ms	151
Thick-PdSe ₂ /thin-PdSe ₂ /Au	0.053	N. A.	5.17×10^{11}	0.72 ms/0.24 ms	154
Thick-Gr/WSe ₂ /Thin-Gr	0.02031	4	4.48×10^9	42.9 μs /56.0 μs	155
Thick-TaSe ₂ /WS ₂ /Thin-TaSe ₂	0.292	57	2.43×10^{11}	43 μs /54 μs	156
Gr/WSe ₂ /NbSe ₂	0.287	88	5.3×10^{11}	80 μs /72 μs	158
Gr/MoS ₂ /N-Gr ^d	0.63	138	7.71×10^{12}	26 μs /50 μs	157
InSe with different contact lengths	0.103	24.5	1.83×10^{10}	1 ms/2.3 ms	167
ReSe ₂ with different contact lengths	0.379	63.49	6.8×10^{11}	18.8 ms/21.6 ms	168
Te with different contact areas	26.1	2090.9	3.24×10^9	720 μs /1.16 ms	169
Gr/PdSe ₂ /Au	$21.6 \text{ V W}^{-1} e$	N. A.	6.7×10^6	42.7 μs /51.9 μs	178
Cr/BP/Au	$\approx 185 \text{ V W}^{-1}$	N. A.	N. A.	0.8 μs /1.5 μs	181
Gr/Sn ₂ S ₂ P ₆ /Gr	3×10^{-3}	0.8	5×10^9	0.23 s/0.19 s	185
Gr/ReS ₂ /Gr	0.11	25.7	N. A.	70.2 μs /79.4 μs	189
Gr/ α -In ₂ Se ₃ /Gr	0.035	8.5	N. A.	N. A.	191
Gr/ α -In ₂ Se ₃ /Gr	1.8×10^{-3}	N. A.	N. A.	N. A.	192
Graphene modified with T-shaped antennas	0.0363	1.1	5×10^6	$\approx 100 \mu\text{s}$	194
Graphene modified with L-shaped antennas	6.3 V W^{-1}	N. A.	N. A.	1 μs	195
Graphene modified with T-shaped antennas	$\approx 1.5 \text{ V W}^{-1}$	N. A.	N. A.	N. A.	196
BSR electrode/Gr/rectangle electrode ^f	138 V W^{-1}	N. A.	N. A.	3.7 μs	197
Graphene with stripe- and triangle-electrodes	0.025	N. A.	N. A.	0.2 ms/0.2 ms	198
PdTe ₂	≈ 0.28	N. A.	N. A.	1 μs /2.2 μs	200
Bi ₂ Se ₃	75	N. A.	N. A.	66 μs /68 μs	201
Sb ₂ Te ₃	0.0385	27.3	3.44×10^{10}	26 μs /20 μs	202

^a N. A.: not applicable. ^b BP: black phosphorus. ^c Gr: graphene. ^d N-Gr: N doped graphene. ^e The output signal is recorded in the form of voltage.

^f BSR: bowtie split ring.

photodetectors is in its nascent stage, and there are still a host of issues that urgently need to be addressed.

(a) Overall, the photosensitivity of the self-driven 2DLM photodetectors based on the new paradigms is still comparatively modest in the current stage. Therefore, elevating device efficiency is pressing task. It is well recognized that this limitation stems predominantly from two disadvantageous ingredients. On one hand, the inherent thinness of 2DLMs significantly curtails the light absorption cross-section, resulting in a substantially limited light absorption cross-section and thereby low absorbance. On the other hand, the separation efficiency of photocarriers is relatively low because the device structures have been far from optimized. In the future, the light–2DLM interactions can be enhanced through strategies such as modifying optical antennas and integrating optical cavities/waveguides. In addition, the impetus for segregating photogenerated electron–hole pairs can be augmented through the simultaneous exertion of multiple strategies (*e.g.* asymmetric contact and thickness tailoring), thereby achieving an improvement in the photoelectric conversion efficiency by leveraging the collaborative effects.

(b) Even though a series of ingenious approaches have been harnessed for self-driven photodetection based on 2DLMs, these strategies, while effective, generally exhibit their own sets of limitations that make them difficult to fully meet the escalating demands for optoelectronic applications necessitating various 2DLMs. For example, self-driven MSM-type photodetectors based on asymmetric electrodes with distinct work functions are in need of the formation of ideal Schottky contacts between the metals and light-sensing channel, which is quite challenging for a variety of 2DLMs due to the complex interfacial effects such as the Fermi level pinning.^{144,204} In addition, self-driven photodetectors based on ferroelectricity-enabled bulk photovoltaic effects entail 2DLMs with low-symmetry crystal structures, which hinders its application in high-symmetry 2DLMs with more mature synthesis process (*e.g.*, MoS₂, WS₂). Furthermore, structural engineering (*e.g.* rolling 2DLMs) requires extremely complex and cumbersome fabrication processes, which can hardly be scalable. In this consideration, it is thus still imperative to develop self-driven 2DLM photodetectors with exceptional working principles. From the preceding advancements, it can be determined that the implementation of self-driven photodetection fundamentally relies on the breaking of symmetry of devices. In addition to thickness and strain, the dielectric environment is also an important parameter affecting the electronic band structure of 2DLMs.^{47,205,206} In this consideration, self-driven photodetection can be theoretically achieved through dielectric contrast engineering. Furthermore, apart from the asymmetry of electronic band structure and electrode, the asymmetry of the light field can also induce self-driven photoresponse by localized heating.^{174,175} Asymmetric light field distribution can be achieved through selective-area modification of plasmonic optical antennas, leading to the development of a new paradigm for self-driven 2DLM photodetectors. By exploiting electrodes of different thicknesses to generate transmittance contrast, another alternative strategy is

afforded for achieving heterogeneous heating.²⁰⁷ In the future, more enthusiasm should be devoted to developing new-paradigm self-driven 2DLM photodetectors.

(c) The large-scale fabrication of ordered arrays of devices stands as a pivotal prerequisite indispensable for the practical exploitation of novel materials in real-world applications. To date, the overwhelming majority of self-driven photodetectors leveraging 2DLMs have been produced through intricate micromechanical exfoliation and transfer methodologies.²⁰⁸ However, these techniques are stalled by inherent limitations in controllability and productivity, coupled with the necessity for cumbersome procedures. Consequently, the quest for scalable preparation of 2DLM-based devices has persisted as a longstanding dilemma. Inspiringly, some pioneering studies have consolidated that synthesis techniques such as interface epitaxy,²⁰⁹ chemical vapor deposition,^{210–213} metal–organic chemical vapor deposition,^{214–216} molecular beam epitaxy,^{213,217,218} pulsed-laser deposition,^{64,219} roll printing,²²⁰ atomic layer deposition,²²¹ and magnetron sputtering^{222,223} boast the potential for scalable preparation. In the future, research endeavors should be focused on refining these large-area growth methodologies and exploring non-destructive patterning techniques, with the aim of preparing diverse 2DLMs in a scalable and efficient manner.

As a concluding remark, there exist a myriad of burgeoning device engineering schemes for the implementation of self-driven photodetection based on 2DLMs. These devices have provided a wide range of avenues for promoting next-generation optoelectronic devices characterized by unparalleled signal-to-noise ratio, low energy consumption, and exceptional integration capabilities, which will catalyze transformative strides in device paradigms. Looking ahead, with further improvements in device performance, the new paradigms of self-driven 2DLM photodetectors are set to shed light on the next era of the optoelectronic industry.

Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this review.

Conflicts of interest

The authors declare no competing interests.

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References

- 1 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666–669.
- 2 J. D. Yao, Z. Q. Zheng and G. W. Yang, *Prog. Mater. Sci.*, 2019, **106**, 100573.
- 3 A. Chaves, J. G. Azadani, H. Alsalman, D. R. da Costa, R. Frisenda, A. J. Chaves, S. H. Song, Y. D. Kim, D. He, J. Zhou, A. Castellanos-Gomez, F. M. Peeters, Z. Liu, C. L. Hinkle, S. H. Oh, P. D. Ye, S. J. Koester, Y. H. Lee, P. Avouris, X. Wang and T. Low, *npj 2D Mater. Appl.*, 2020, **4**, 29.
- 4 J. Yao and G. Yang, *Nanoscale*, 2020, **12**, 454–476.
- 5 Q. Fu, J. Han, X. Wang, P. Xu, T. Yao, J. Zhong, W. Zhong, S. Liu, T. Gao, Z. Zhang, L. Xu and B. Song, *Adv. Mater.*, 2021, **33**, 1907818.
- 6 J. Yao and G. Yang, *Nanotechnology*, 2021, **32**, 392001.
- 7 S. Das, A. Sebastian, E. Pop, C. J. McClellan, A. D. Franklin, T. Grasser, T. Knobloch, Y. Illarionov, A. V. Penumatcha, J. Appenzeller, Z. Chen, W. Zhu, I. Asselberghs, L.-J. Li, U. E. Avci, N. Bhat, T. D. Anthopoulos and R. Singh, *Nat. Electron.*, 2021, **4**, 786–799.
- 8 J. F. Sierra, J. Fabian, R. K. Kawakami, S. Roche and S. O. Valenzuela, *Nat. Nanotechnol.*, 2021, **16**, 856–868.
- 9 P. Ares and K. S. Novoselov, *Nano Mater. Sci.*, 2022, **4**, 3–9.
- 10 Q. H. Wang, A. Bedoya-Pinto, M. Blei, A. H. Dismukes, A. Hamo, S. Jenkins, M. Koperski, Y. Liu, Q. C. Sun, E. J. Telford, H. H. Kim, M. Augustin, U. Vool, J. X. Yin, L. H. Li, A. Falin, C. R. Dean, F. Casanova, R. F. L. Evans, M. Chshiev, A. Mishchenko, C. Petrovic, R. He, L. Zhao, A. W. Tsen, B. D. Gerardot, M. Brotons-Gisbert, Z. Guguchia, X. Roy, S. Tongay, Z. Wang, M. Z. Hasan, J. Wrachtrup, A. Yacoby, A. Fert, S. Parkin, K. S. Novoselov, P. Dai, L. Balicas and E. J. G. Santos, *ACS Nano*, 2022, **16**, 6960–7079.
- 11 J. Qi, Z. Wu, W. Wang, K. Bao, L. Wang, J. Wu, C. Ke, Y. Xu and Q. He, *Int. J. Extreme Manuf.*, 2023, **5**, 022007.
- 12 S. Song, M. Rahaman and D. Jariwala, *ACS Nano*, 2024, **18**, 10955–10978.
- 13 Z. Huang, Y. Li, Y. Zhang, J. Chen, J. He and J. Jiang, *Int. J. Extreme Manuf.*, 2024, **6**, 032003.
- 14 R. Wu, H. Zhang, H. Ma, B. Zhao, W. Li, Y. Chen, J. Liu, J. Liang, Q. Qin, W. Qi, L. Chen, J. Li, B. Li and X. Duan, *Chem. Rev.*, 2024, **124**, 10112–10191.
- 15 D. Zeng, Z. Zhang, Z. Xue, M. Zhang, P. K. Chu, Y. Mei, Z. Tian and Z. Di, *Nature*, 2024, **632**, 788–794.
- 16 Q. Deng, T. Zhao, J. Zhang, W. Yue, L. Li, S. Li, L. Zhu, Y. Sun, Y. Pan, T. Zheng, X. Liu, Y. Yan and N. Huo, *ACS Nano*, 2024, **18**, 23702–23710.
- 17 R. Pendurthi, N. U. Sakib, M. U. K. Sadaf, Z. Zhang, Y. Sun, C. Chen, D. Jayachandran, A. Oberoi, S. Ghosh, S. Kumari, S. P. Stepanoff, D. Somvanshi, Y. Yang, J. M. Redwing, D. E. Wolfe and S. Das, *Nat. Nanotechnol.*, 2024, **19**, 970–977.
- 18 L. Liu, Y. Sun, X. Huang, C. Liu, Z. Tang, S. Zeng, D. W. Zhang, S. Deng and P. Zhou, *Mater. Futures*, 2022, **1**, 025301.
- 19 M. Nonahal, C. Li, H. Ren, L. Spencer, M. Kianinia, M. Toth and I. Aharonovich, *Laser Photonics Rev.*, 2023, **17**, 2300019.
- 20 M. O. Ramírez, P. Molina, D. Hernández-Pinilla, G. López-Polín, P. Ares, L. Lozano-Martín, H. Yan, Y. Wang, S. Sarkar, J. H. Al Shuhaib, F. Leardini, J. Gómez-Herrero, M. Chhowalla and L. E. Bausá, *Laser Photonics Rev.*, 2024, **18**, 2300817.
- 21 H. Tang, Y. Wang, X. Ni, K. Watanabe, T. Taniguchi, P. Jarillo-Herrero, S. Fan, E. Mazur, A. Yacoby and Y. Cao, *Nature*, 2024, **632**, 1038–1044.
- 22 Q. Liu, W. Xu, X. Li, T. Zhang, C. Qin, F. Luo, Z. Zhu, S. Qin, M. Zhu and K. S. Novoselov, *Int. J. Extreme Manuf.*, 2024, **6**, 015501.
- 23 J. P. Wang, X. Chen, Q. Zhao, Y. Fang, Q. Liu, J. Fu, Y. Liu, X. Xu, J. Zhang, L. Zhen, C. Y. Xu, F. Huang, A. J. Meixner, D. Zhang, G. Gou and Y. Li, *ACS Nano*, 2024, **18**, 16274–16284.
- 24 H. Cai, A. Rasmita, R. He, Z. Zhang, Q. Tan, D. Chen, N. Wang, Z. Mu, J. J. H. Eng, Y. She, N. Pan, Q. Wang, Z. Dong, X. Wang, J. Wang, Y. Miao, R. Singh, C. W. Qiu, X. Liu and W. Gao, *Nat. Photonics*, 2024, **18**, 842–847.
- 25 Z. Wang, V. Kalathingai, M. Trushin, J. Liu, J. Wang, Y. Guo, B. Özyilmaz, C. A. Nijhuis and G. Eda, *Nat. Nanotechnol.*, 2024, **19**, 993–999.
- 26 Z. He, W. Chen, K. Hu, Y. Luo, W. Zeng, X. He, T. Li, J. Ouyang, Y. Li, L. Xie, Y. Zhang, Q. Xu, S. Yang, M. Guo, W. Zou, Y. Li, L. Huang, L. Chen, X. Zhang, Q. Saiding, R. Wang, M. R. Zhang, N. Kong, T. Xie, X. Song and W. Tao, *Nat. Nanotechnol.*, 2024, **19**, 1386–1398.
- 27 J. Xu, X. Yuan, Y. Zhao, S. Rui, Q. Jia, H. Li, S. Lu, B. Li, Y. Zhang and X. Zhu, *Prog. Nat. Sci.: Mater. Int.*, 2024, **34**, 429–436.
- 28 Y. L. Wu, J. B. Hong, W. X. Zhong, C. X. Wang, Z. F. Li and S. Dmytro, *Tungsten*, 2024, **6**, 124–133.
- 29 X. Wen, W. Feng, X. Li, J. Yang, R. Du, P. Wang, H. Li, L. Song, Y. Wang, M. Cheng, J. He and J. Shi, *Adv. Mater.*, 2023, **35**, 2211690.
- 30 H. Y. Zou, L. Fang, G. Yu and D. Wang, *Tungsten*, 2024, **6**, 248–258.
- 31 Y. Lin, P. Xinxin, W. Zhenhai and H. Haitao, *Mater. Futures*, 2022, **1**, 035103.
- 32 Y. Zhai, H. He, M. Xu, H. Zhao and J. Li, *Prog. Nat. Sci.: Mater. Int.*, 2023, **33**, 442–449.
- 33 J. Yansong, S. Guining, L. Yang, Y. Ruizhe, H. Ming, H. Hua, L. Min, H. Gai, L. Qunjie and G. Chaohua, *Microstructures*, 2024, **4**, 2024038.
- 34 L. Zhang, Z. Yang, S. Feng, Z. Guo, Q. Jia, H. Zeng, Y. Ding, P. Das, Z. Bi, J. Ma, Y. Fu, S. Wang, J. Mi, S. Zheng, M. Li, D. M. Sun, N. Kang, Z. S. Wu and H. M. Cheng, *Nature*, 2024, **628**, 313–319.
- 35 M. Chen, X. Chen, Z. Wu, Z. Huang, W. Gao, M. Yang, Y. Xiao, Y. Zhao, Z. Zheng, J. Yao and J. Li, *ACS Nano*, 2024, **18**, 27579–27589.

- 36 Z. Zhou, F. Hou, X. Huang, G. Wang, Z. Fu, W. Liu, G. Yuan, X. Xi, J. Xu, J. Lin and L. Gao, *Nature*, 2023, **621**, 499–505.
- 37 X. B. Shi, P. He and W. W. Zhao, *Tungsten*, 2023, **5**, 357–363.
- 38 C. Li, F. Xu, B. Li, J. Li, G. Li, K. Watanabe, T. Taniguchi, B. Tong, J. Shen, L. Lu, J. Jia, F. Wu, X. Liu and T. Li, *Nature*, 2024, **631**, 300–306.
- 39 J. Yao and G. Yang, *Adv. Sci.*, 2022, **9**, 2103036.
- 40 Y. Pinki, S. Dewan, R. Mishra and S. Das, *J. Phys. D: Appl. Phys.*, 2022, **55**, 313001.
- 41 J. Yao and G. Yang, *J. Appl. Phys.*, 2022, **131**, 161101.
- 42 P. P. Satheesh, H. S. Jang, B. Pandit, S. Chandramohan and K. Heo, *Adv. Funct. Mater.*, 2023, **33**, 2212167.
- 43 M. Dai, X. Zhang and Q. J. Wang, *Adv. Funct. Mater.*, 2024, **34**, 2312872.
- 44 Y. Wang, L. Mei, Y. Li, X. Xia, N. Cui, G. Long, W. Yu, W. Chen, H. Mu and S. Lin, *Phys. Rep.*, 2024, **1081**, 1–46.
- 45 Y. Ma, H. Yi, H. Liang, W. Wang, Z. Zheng, J. Yao and G. Yang, *Mater. Futures*, 2024, **3**, 012301.
- 46 Y. Ma, H. Liang, X. Guan, S. Xu, M. Tao, X. Liu, Z. Zheng, J. Yao and G. Yang, *Nanoscale Horiz.*, 2024, **9**, 1599–1629.
- 47 J. Lu, Z. Deng, Q. Ye, Z. Zheng, J. Yao and G. Yang, *Small Methods*, 2022, **6**, 2101046.
- 48 Y. Zou, Z. Zhang, J. Yan, L. Lin, G. Huang, Y. Tan, Z. You and P. Li, *Nat. Commun.*, 2022, **13**, 4372.
- 49 Q. Ye, D. Xu, B. Cai, J. Lu, H. Yi, C. Ma, Z. Zheng, J. Yao, G. Ouyang and G. Yang, *Mater. Horiz.*, 2022, **9**, 2364–2375.
- 50 Q. Ye, J. Lu, Z. Zheng, W. Huang, J. Yao and G. Yang, *Adv. Opt. Mater.*, 2022, **10**, 2102335.
- 51 Y. Yu, M. Peng, F. Zhong, Z. Wang, X. Ge, H. Chen, J. Guo, Y. Wang, Y. Chen, T. Xu, T. Zhao, T. He, K. Zhang, F. Wu, C. Chen, J. Dai and W. Hu, *Mater. Horiz.*, 2023, **10**, 2579–2586.
- 52 Q. Ye, Z. Deng, H. Yi, W. Wang, J. Lu, Y. Ma, W. Huang, Z. Zheng, C. Ma, C. Du, Y. Zou, J. Yao and G. Yang, *Adv. Opt. Mater.*, 2023, **11**, 2300463.
- 53 J. Lu, Y. He, C. Ma, Q. Ye, H. Yi, Z. Zheng, J. Yao and G. Yang, *Adv. Mater.*, 2023, **35**, 2211562.
- 54 W. R. Syong, J. H. Fu, Y. H. Kuo, Y. C. Chu, M. Hakami, T. Y. Peng, J. Lynch, D. Jariwala, V. Tung and Y. J. Lu, *ACS Nano*, 2024, **18**, 5446–5456.
- 55 Q. Wu, C. Wang, L. Li, X. Zhang, Y. Jiang, Z. Cai, L. Lin, Z. Ni, X. Gu, K. Ostrikov, H. Nan and S. Xiao, *J. Mater. Sci. Technol.*, 2024, **193**, 217–225.
- 56 Y. Hu, J. He, Z. Yan, C. Xu, X. Li, N. Wei, Y. Wang, N. Dong and J. Wang, *Laser Photonics Rev.*, 2024, 2400661.
- 57 X. Liu, J. Zhou, Z. Jiang, Z. Huang, B. Li, Z. Ma, Y. Yang, Y. Huang, Y. Zhang, V. D. Botcha, R. J. Chung, J. Liang, X. Li, Y. Li and W. He, *Adv. Electron. Mater.*, 2024, **10**, 2300711.
- 58 H. Jiang, J. Fu, J. Wei, S. Li, C. Nie, F. Sun, Q. Y. S. Wu, M. Liu, Z. Dong, X. Wei, W. Gao and C. W. Qiu, *Nat. Commun.*, 2024, **15**, 1225.
- 59 X. L. Zhang, J. Li, B. Leng, L. Yang, Y. D. Song, S. Y. Feng, L. Z. Feng, Z. T. Liu, Z. W. Fu, X. Jiang and B. D. Liu, *Tungsten*, 2023, **5**, 91–99.
- 60 X. Yang, C. J. Wang, S. Cheng, X. G. Yang, J. H. Zang and C. X. Shan, *Microstructures*, 2023, **3**, 2023005.
- 61 L. Zhang, Y. He, X. Dong, J. Guo, Z. Gao, Z. Liu, J. Chen, Y. Zhao, Z. Zhou, J. Yin, X. Fu, F. Luo, H. Fu and J. Wu, *Laser Photonics Rev.*, 2024, **18**, 2300854.
- 62 H. Guan, F. Liu, Z. He, H. Xie, M. Xie, Z. Fang, M. Yang, B. Chen, X. Liang, F. Li, Y. Wei, T. Yang and H. Lu, *Laser Photonics Rev.*, 2024, 2400445.
- 63 L. Chen, H. Ma, C. Liu, D. Wang, Z. Han, J. Tai, H. Liang and H. Yin, *Laser Photonics Rev.*, 2024, 2400304.
- 64 W. Wang, X. Wang, J. Yao, H. Chen and G. Yang, *Adv. Opt. Mater.*, 2024, **12**, 2401937.
- 65 H. Zhao, X. Guo, Y. Wang, W. Wang, S. Li, Q. Zhou, T. Zhou, Y. Xie, Y. Yu, F. Xuan, Z. Ni and L. Gao, *Laser Photonics Rev.*, 2024, 2400192.
- 66 Q. Wu, J. Qian, Y. Wang, L. Xing, Z. Wei, X. Gao, Y. Li, Z. Liu, H. Liu, H. Shu, J. Yin, X. Wang and H. Peng, *Nat. Commun.*, 2024, **15**, 3688.
- 67 S. Debnath, K. Ghosh, M. Meyyappan and P. K. Giri, *Nanoscale*, 2023, **15**, 13809–13821.
- 68 S. Li, X. Liu, H. Yang, H. Zhu and X. Fang, *Nat. Electron.*, 2024, **7**, 216–224.
- 69 H. S. Lee, J. Ahn, W. Shim, S. Im and D. K. Hwang, *Appl. Phys. Lett.*, 2018, **113**, 163102.
- 70 Z. Zou, J. Liang, X. Zhang, C. Ma, P. Xu, X. Yang, Z. Zeng, X. Sun, C. Zhu, D. Liang, X. Zhuang, D. Li and A. Pan, *ACS Nano*, 2021, **15**, 10039–10047.
- 71 T. Zhu, K. Liu, Y. Zhang, S. Meng, M. He, Y. Zhang, M. Yan, X. Dong, X. Li, M. Jiang and H. Xu, *ACS Nano*, 2024, **18**, 11462–11473.
- 72 T. Xu, F. Zhong, P. Wang, Z. Wang, X. Ge, J. Wang, H. Wang, K. Zhang, Z. Zhang, T. Zhao, Y. Yu, M. Luo, Y. Wang, R. Jiang, F. Wang, F. Chen, Q. Liu and W. Hu, *Sci. Adv.*, 2024, **10**, eadn0560.
- 73 F. Gong, H. Fang, P. Wang, M. Su, Q. Li, J. C. Ho, X. Chen, W. Lu, L. Liao, J. Wang and W. Hu, *Nanotechnology*, 2017, **28**, 484002.
- 74 D. Wu, M. Xu, L. Zeng, Z. Shi, Y. Tian, X. J. Li, C. X. Shan and J. Jie, *ACS Nano*, 2022, **16**, 5545–5555.
- 75 D. Wu, C. Guo, L. Zeng, X. Ren, Z. Shi, L. Wen, Q. Chen, M. Zhang, X. J. Li, C. X. Shan and J. Jie, *Light: Sci. Appl.*, 2023, **12**, 5.
- 76 C. Zhou, S. Raju, B. Li, M. Chan, Y. Chai and C. Y. Yang, *Adv. Funct. Mater.*, 2018, **28**, 1802954.
- 77 D. Lu, Y. Chen, L. Kong, C. Luo, Z. Lu, Q. Tao, W. Song, L. Ma, Z. Li, W. Li, L. Liu, Q. Li, X. Yang, J. Li, J. Li, X. Duan, L. Liao and Y. Liu, *Small*, 2022, **18**, 2107104.
- 78 Y. Hu, J. Wang, M. Tamtaji, Y. Feng, T. W. Tang, M. Amjadian, T. Kang, M. Xu, X. Shi, D. Zhao, Y. Mi, Z. Luo and L. An, *Adv. Mater.*, 2024, 2404013.
- 79 R. Maiti, C. Patil, M. A. S. R. Saadi, T. Xie, J. G. Azadani, B. Uluotku, R. Amin, A. F. Briggs, M. Miscuglio, D. Van Thourhout, S. D. Solares, T. Low, R. Agarwal, S. R. Bank and V. J. Sorger, *Nat. Photonics*, 2020, **14**, 578–584.

- 80 H. Yi, C. Ma, W. Wang, H. Liang, R. Cui, W. Cao, H. Yang, Y. Ma, W. Huang, Z. Zheng, Y. Zou, Z. Deng, J. Yao and G. Yang, *Mater. Horiz.*, 2023, **10**, 3369–3381.
- 81 Y. Zhou, L. Han, Q. Song, W. Gao, M. Yang, Z. Zheng, L. Huang, J. Yao and J. Li, *Sci. China Mater.*, 2022, **65**, 732–740.
- 82 M. Yang, W. Gao, Q. Song, Y. Zhou, L. Huang, Z. Zheng, Y. Zhao, J. Yao and J. Li, *Adv. Opt. Mater.*, 2021, **9**, 2100450.
- 83 S. Q. Zhang, X. L. Zou and H. M. Cheng, *Sci. China: Technol. Sci.*, 2020, **63**, 1265–1271.
- 84 S. B. Chen, Z. Y. Zeng, X. R. Chen and X. X. Yao, *J. Appl. Phys.*, 2020, **128**, 125111.
- 85 J. Zhou, T. Y. Cai and S. Ju, *Phys. Rev. Appl.*, 2021, **15**, 024045.
- 86 H. Li, G. Sanchez-Santolino, S. Puebla, R. Frisenda, A. M. Al-Enizi, A. Nafady, R. D'Agosta and A. Castellanos-Gomez, *Adv. Mater.*, 2022, **34**, 2103571.
- 87 L. Zhao, Y. Jiang, C. Li, Y. Liang, Z. Wei, X. Wei and Q. Zhang, *Nano Lett.*, 2023, **23**, 3493–3500.
- 88 H. Wang, S. Liu, Z. Liu, Y. Sun, D. Xie and T. Ren, *Adv. Mater.*, 2024, **36**, 2305770.
- 89 Z. Li, Y. Lv, L. Ren, J. Li, L. Kong, Y. Zeng, Q. Tao, R. Wu, H. Ma, B. Zhao, D. Wang, W. Dang, K. Chen, L. Liao, X. Duan, X. Duan and Y. Liu, *Nat. Commun.*, 2020, **11**, 1151.
- 90 Y. Kang, Y. Pei, D. He, H. Xu, M. Ma, J. Yan, C. Jiang, W. Li and X. Xiao, *Light: Sci. Appl.*, 2024, **13**, 127.
- 91 M. Kang, W. Hong, I. Lee, S. Park, C. Park, S. Bae, H. Lim and S. Y. Choi, *ACS Appl. Mater. Interfaces*, 2024, **16**, 43849–43859.
- 92 J. Sun, Y. Wang, S. Guo, B. Wan, L. Dong, Y. Gu, C. Song, C. Pan, Q. Zhang, L. Gu, F. Pan and J. Zhang, *Adv. Mater.*, 2020, **32**, 1906499.
- 93 X. Fan, S. He, P. Feng, Y. Xiao, C. Yin, Y. A. Du, M. Li, L. Zhao and L. Gao, *J. Phys. Chem. Lett.*, 2024, **15**, 5923–5934.
- 94 Y. Lu, Z. Zhan, J. Tan, H. Lai, P. Liu, Y. Zhou and W. Xie, *Laser Photonics Rev.*, 2024, **18**, 2300941.
- 95 D. J. Groenendijk, M. Buscema, G. A. Steele, S. Michaelis de Vasconcellos, R. Bratschitsch, H. S. J. van der Zant and A. Castellanos-Gomez, *Nano Lett.*, 2014, **14**, 5846–5852.
- 96 A. Pospischil, M. M. Furchi and T. Mueller, *Nat. Nanotechnol.*, 2014, **9**, 257–261.
- 97 B. W. H. Baugher, H. O. H. Churchill, Y. Yang and P. Jarillo-Herrero, *Nat. Nanotechnol.*, 2014, **9**, 262–267.
- 98 S. Memaran, N. R. Pradhan, Z. Lu, D. Rhodes, J. Ludwig, Q. Zhou, O. Ogunsolu, P. M. Ajayan, D. Smirnov, A. I. Fernández-Domínguez, F. J. García-Vidal and L. Balicas, *Nano Lett.*, 2015, **15**, 7532–7538.
- 99 M. Buscema, D. J. Groenendijk, G. A. Steele, H. S. J. van der Zant and A. Castellanos-Gomez, *Nat. Commun.*, 2014, **5**, 4651.
- 100 Y. Q. Bie, G. Grosso, M. Heuck, M. M. Furchi, Y. Cao, J. Zheng, D. Bunandar, E. Navarro Moratalla, L. Zhou, D. K. Efetov, T. Taniguchi, K. Watanabe, J. Kong, D. Englund and P. Jarillo Herrero, *Nat. Nanotechnol.*, 2017, **12**, 1124–1129.
- 101 Y. Zhang, X. Chen, M. Zhang, X. Wu, J. Wang, R. Tian, L. Fang, Y. Zhang, J. Zhao and X. Gan, *Adv. Funct. Mater.*, 2024, **34**, 2402957.
- 102 H. Kim and H. J. Choi, *Phys. Rev. B*, 2021, **103**, 085404.
- 103 C. Wang, Q. Wu, Y. Ding, Z. Cai, S. Xiao, X. Zhang, H. Nan and X. Gu, *Appl. Phys. Express*, 2023, **16**, 015003.
- 104 X. Zhu, W. Niu, H. Fu, J. Miao, H. Huang, X. He, X. Liu, X. Zou, F. Shan and Z. Yang, *IEEE Trans. Electron Devices*, 2024, **71**, 2192–2195.
- 105 Y. Yang, N. Huo and J. Li, *J. Mater. Chem. C*, 2017, **5**, 7051–7056.
- 106 Z. Wang, Y. Chen, P. Wu, J. Ye, M. Peng, Y. Yan, F. Zhong, T. He, Y. Wang, M. Xu, K. Zhang, Z. Hu, Q. Li, L. Zhang, F. Wang and P. Wang, *Infrared Phys. Technol.*, 2020, **106**, 103272.
- 107 C. Tan, H. Wang, X. Zhu, W. Gao, H. Li, J. Chen, G. Li, L. Chen, J. Xu, X. Hu, L. Li and T. Zhai, *ACS Appl. Mater. Interfaces*, 2020, **12**, 44934–44942.
- 108 S. Park, J. Ha, M. F. Khan, C. Im, J. Y. Park, S. H. Yoo, M. A. Rehman, K. Kang, S. H. Lee and S. C. Jun, *ACS Appl. Electron. Mater.*, 2022, **4**, 4306–4315.
- 109 J. Sun, G. Giorgi, M. Palummo, P. Sutter, M. Passacantando and L. Camilli, *ACS Nano*, 2020, **14**, 4861–4870.
- 110 P. Luo, F. Wang, J. Qu, K. Liu, X. Hu, K. Liu and T. Zhai, *Adv. Funct. Mater.*, 2021, **31**, 2008351.
- 111 L. Zhang, X. Han, S. Zhang, H. Wang, Y. Huang, Z. Zheng, N. Huo, W. Gao and J. Li, *Adv. Electron. Mater.*, 2022, **8**, 2200551.
- 112 J. Wang, C. Fu, M. Jiang, Y. Hu, Y. Liu, M. L. Zhu, J. Yu, J. Fu, R. Lin, D. Wu, Z. Mahfoud, S. A. Jia, F. X. Liang, L. Li, J. Teng and L. B. Luo, *Adv. Opt. Mater.*, 2024, 2401035.
- 113 X. Wang, S. Wang, Y. Wu, W. Wang, Z. Cao, B. Wei, T. Han, F. Li, S. Wang, L. Shan and M. Long, *Adv. Opt. Mater.*, 2024, **12**, 2400052.
- 114 Z. Fu, J. Han, L. Wei, Y. Wei, X. Huo, C. Li, J. Xie, H. Zhou, H. Yu, J. Gou, Z. Wu and J. Wang, *Adv. Opt. Mater.*, 2024, 2401306.
- 115 Z. Xu, Q. Qin, X. Ma, J. Chen, X. Liu, W. Chen, Z. Qiu, L. Wu, W. Gao and L. Li, *Adv. Opt. Mater.*, 2024, 2401379.
- 116 J. You, Z. Jin, Y. Li, T. Kang, K. Zhang, W. Wang, M. Xu, Z. Gao, J. Wang, J. K. Kim and Z. Luo, *Adv. Funct. Mater.*, 2024, **34**, 2311134.
- 117 B. Zhang, Z. Ao, X. Lan, J. Zhong, F. Zhang, S. Zhang, R. Yang, L. Wang, P. Chen, G. Wang, X. Yang, H. Liu, J. Cao, M. Zhong, H. Li and Z. Zhang, *Nano Lett.*, 2024, **24**, 7716–7723.
- 118 B. Zhao, Z. Wan, Y. Liu, J. Xu, X. Yang, D. Shen, Z. Zhang, C. Guo, Q. Qian, J. Li, R. Wu, Z. Lin, X. Yan, B. Li, Z. Zhang, H. Ma, B. Li, X. Chen, Y. Qiao, I. Shakir, Z. Almutairi, F. Wei, Y. Zhang, X. Pan, Y. Huang, Y. Ping, X. Duan and X. Duan, *Nature*, 2021, **591**, 385–390.

- 119 X. Cui, Z. Kong, E. Gao, D. Huang, Y. Hao, H. Shen, C. Di, Z. Xu, J. Zheng and D. Zhu, *Nat. Commun.*, 2018, **9**, 1301.
- 120 G. Wu, B. Tian, L. Liu, W. Lv, S. Wu, X. Wang, Y. Chen, J. Li, Z. Wang, S. Wu, H. Shen, T. Lin, P. Zhou, Q. Liu, C. Duan, S. Zhang, X. Meng, S. Wu, W. Hu, X. Wang, J. Chu and J. Wang, *Nat. Electron.*, 2020, **3**, 43–50.
- 121 G. Wu, X. Wang, Y. Chen, S. Wu, B. Wu, Y. Jiang, H. Shen, T. Lin, Q. Liu, X. Wang, P. Zhou, S. Zhang, W. Hu, X. Meng, J. Chu and J. Wang, *Adv. Mater.*, 2020, **32**, 1907937.
- 122 L. Lv, F. Zhuge, F. Xie, X. Xiong, Q. Zhang, N. Zhang, Y. Huang and T. Zhai, *Nat. Commun.*, 2019, **10**, 3331.
- 123 M. Zhang, G. Zeng, G. Wu, J. Zeng, Y. Sun, C. Li, L. Liu, J. Wang, H.-L. Lu, Y. Chai and J. Wang, *Appl. Phys. Lett.*, 2023, **122**, 253503.
- 124 S. Wu, Y. Chen, X. Wang, H. Jiao, Q. Zhao, X. Huang, X. Tai, Y. Zhou, H. Chen, X. Wang, S. Huang, H. Yan, T. Lin, H. Shen, W. Hu, X. Meng, J. Chu and J. Wang, *Nat. Commun.*, 2022, **13**, 3198.
- 125 G. Wu, X. Wang, Y. Chen, S. Wu, H. Shen, T. Lin, J. Ge, W. Hu, S. T. Zhang, X. J. Meng, J. Chu and J. Wang, *Appl. Phys. Lett.*, 2020, **116**, 073101.
- 126 F. Guo, W. F. Io, Z. Dang, R. Ding, S. Y. Pang, Y. Zhao and J. Hao, *Mater. Horiz.*, 2023, **10**, 3719–3728.
- 127 W. F. Io, S. Y. Pang, L. W. Wong, Y. Zhao, R. Ding, J. Mao, Y. Zhao, F. Guo, S. Yuan, J. Zhao, J. Yi and J. Hao, *Nat. Commun.*, 2023, **14**, 7304.
- 128 J. Han, C. Fang, M. Yu, J. Cao and K. Huang, *Adv. Electron. Mater.*, 2022, **8**, 2100987.
- 129 T. Mueller, F. Xia and P. Avouris, *Nat. Photonics*, 2010, **4**, 297–301.
- 130 Y. Liu, J. Guo, E. Zhu, L. Liao, S. J. Lee, M. Ding, I. Shakir, V. Gambin, Y. Huang and X. Duan, *Nature*, 2018, **557**, 696–700.
- 131 M. Dai, H. Chen, R. Feng, W. Feng, Y. Hu, H. Yang, G. Liu, X. Chen, J. Zhang, C. Y. Xu and P. Hu, *ACS Nano*, 2018, **12**, 8739–8747.
- 132 Y. Mao, P. Xu, Q. Wu, J. Xiong, R. Peng, W. Huang, S. Chen, Z. Wu and C. Li, *Adv. Electron. Mater.*, 2021, **7**, 2001138.
- 133 T. Y. Wu, L. W. Wang, B. C. Chen, J. H. Wang and S. Y. Chu, *ACS Appl. Electron. Mater.*, 2024, **6**, 702–711.
- 134 J. Park, S. Kim, M. Yang, H. Hosono, K. Park, J. Yoon, J. Bak, B. You, S. W. Park, M. G. Hahm and M. Lee, *ACS Photonics*, 2023, **10**, 2930–2940.
- 135 Z. Kang, Y. Cheng, Z. Zheng, F. Cheng, Z. Chen, L. Li, X. Tan, L. Xiong, T. Zhai and Y. Gao, *Nano-Micro Lett.*, 2019, **11**, 34.
- 136 A. K. Mia, M. Meyyappan and P. K. Giri, *Nanoscale*, 2024, **16**, 8583–8596.
- 137 T. Kang, J. You, J. Wang, Y. Li, Y. Hu, T. W. Tang, X. Lin, Y. Li, L. Liu, Z. Gao, Y. Liu and Z. Luo, *Nano Lett.*, 2024, **24**, 8369–8377.
- 138 G. Wang, F. Liu, R. Chen, M. Wang, Y. Yin, J. Zhang, Z. Sa, P. Li, J. Wan, L. Sun, Z. Lv, Y. Tan, F. Chen and Z. Yang, *Small*, 2024, **20**, 2306363.
- 139 J. Tai, L. Chen, D. Wang, W. Gao, Z. Long, H. Wang, H. Liang and H. Yin, *J. Mater. Sci. Technol.*, 2025, **217**, 128–137.
- 140 S. Xing, S. Chen, S. Fang, F. Tian, Z. Li, X. Jin, Q. Li and B. Liu, *Adv. Opt. Mater.*, 2024, 2401433.
- 141 M. Dai, Q. Wu, C. Wang, X. Liu, X. Zhang, Z. Cai, L. Lin, X. Gu, K. Ostrikov, H. Nan and S. Xiao, *Adv. Opt. Mater.*, 2024, **12**, 2301900.
- 142 G. Wang, Y. Sun, Z. Yang, W. Lu, S. Chen, X. Zhang, H. Ma, T. Sun, P. Huo, X. Cui, B. Man, X. Wang and C. Yang, *Adv. Funct. Mater.*, 2024, **34**, 2316267.
- 143 C. Zhou, S. Zhang, Z. Lv, Z. Ma, C. Yu, Z. Feng and M. Chan, *npj 2D Mater. Appl.*, 2020, **4**, 46.
- 144 X. Zhang, B. Liu, L. Gao, H. Yu, X. Liu, J. Du, J. Xiao, Y. Liu, L. Gu, Q. Liao, Z. Kang, Z. Zhang and Y. Zhang, *Nat. Commun.*, 2021, **12**, 1522.
- 145 H. Wang, W. Wang, Y. Zhong, D. Li, Z. Li, X. Xu, X. Song, Y. Chen, P. Huang, A. Mei, H. Han, T. Zhai and X. Zhou, *Adv. Mater.*, 2022, **34**, 2206122.
- 146 N. Bu, J. Huang, S. Chen, L. Zhu, H. Yu, L. Li, J. Li and N. Huo, *Appl. Phys. Lett.*, 2023, **123**, 033302.
- 147 M. Hussain, S. Aftab, S. H. A. Jaffery, A. Ali, S. Hussain, D. N. Cong, R. Akhtar, Y. Seo, J. Eom, P. Gautam, H. Noh and J. Jung, *Sci. Rep.*, 2020, **10**, 9374.
- 148 J. Huang, K. Shu, N. Bu, Y. Yan, T. Zheng, M. Yang, Z. Zheng, N. Huo, J. Li and W. Gao, *Sci. China Mater.*, 2023, **66**, 4711–4722.
- 149 K. Nassiri Nazif, A. Daus, J. Hong, N. Lee, S. Vaziri, A. Kumar, F. Nitta, M. E. Chen, S. Kananian, R. Islam, K. H. Kim, J. H. Park, A. S. Y. Poon, M. L. Brongersma, E. Pop and K. C. Saraswat, *Nat. Commun.*, 2021, **12**, 7034.
- 150 K. Tang, C. Yan, X. Du, G. Rao, M. Zhang, Y. Wang, X. Wang and J. Xiong, *Adv. Opt. Mater.*, 2024, **12**, 2301350.
- 151 C. Yan, Y. Li, R. Li, R. Ma, J. Li, W. Xin, W. Liu, H. Xu and Y. Liu, *Laser Photonics Rev.*, 2024, 2400951.
- 152 S. Ghods, H. Lee, J. H. Choi, J. Y. Moon, S. Kim, S.-I. Kim, H. J. Kwun, M. J. Josline, C. Y. Kim, S. H. Hyun, S. W. Kim, S. K. Son, T. Lee, Y. K. Lee, K. Heo, K. S. Novoselov and J. H. Lee, *ACS Nano*, 2024, **18**, 26192–26200.
- 153 L. Gao, B. Kang, H. Wu and S. Jiang, *AIP Adv.*, 2024, **14**, 045029.
- 154 X. Zhang, M. Dai, W. Deng, Y. Zhang and Q. J. Wang, *Nanophotonics*, 2023, **12**, 607–618.
- 155 L. Tong, C. Su, H. Li, X. Wang, W. Fan, Q. Wang, S. Kunsági-Máté, H. Yan and S. Yin, *ACS Appl. Mater. Interfaces*, 2023, **15**, 57868–57878.
- 156 X. Wang, L. Tong, W. Fan, W. Yan, C. Su, D. Wang, Q. Wang, H. Yan and S. Yin, *J. Colloid Interface Sci.*, 2024, **657**, 529–537.
- 157 J. X. Guo, S. D. Li, L. Lin, J. Cai, J. B. Chen, S. C. Wang, X. Gou, J. H. Ye, Z. H. Luo and W. Huang, *Sci. China: Technol. Sci.*, 2022, **65**, 2658–2666.
- 158 S. He, C. Yin, L. Zhang, Y. Chen, H. Peng, A. Shan, L. Zhao and L. Gao, *J. Mater. Sci. Technol.*, 2024, DOI: [10.1016/j.jmst.2024.1008.1055](https://doi.org/10.1016/j.jmst.2024.1008.1055).

- 159 S. Liu, Y. Chen, S. Yang and C. Jiang, *Nano Res.*, 2022, **15**, 1532–1537.
- 160 C. S. Lee, S. J. Oh, H. Heo, S. Y. Seo, J. Kim, Y. H. Kim, D. Kim, O. F. Ngome Okello, H. Shin, J. H. Sung, S. Y. Choi, J. S. Kim, J. K. Kim and M. H. Jo, *Nano Lett.*, 2019, **19**, 1814–1820.
- 161 H. Liu, L. Fang, X. Zhu, C. Zhu, X. Sun, G. Xu, B. Zheng, Y. Liu, Z. Luo, H. Wang, C. Yao, D. Li and A. Pan, *Nano Res.*, 2023, **16**, 11832–11838.
- 162 Y. Wang, C. Liu, H. Duan, Z. Li, C. Wang, H. Tan, S. Feng, R. Liu, P. Li and W. Yan, *Nanotechnology*, 2024, **35**, 025601.
- 163 S. S. Han, S. Sattar, D. Kireev, J. C. Shin, T. S. Bae, H. I. Ryu, J. Cao, A. K. Shum, J. H. Kim, C. M. Canali, D. Akinwande, G. H. Lee, H. S. Chung and Y. Jung, *Nano Lett.*, 2024, **24**, 1891–1900.
- 164 B. Wei, Y. Li, T. Yun, Y. Li, T. Gui, W. Yu, H. Mu, N. Cui, W. Chen and S. Lin, *Mater. Futures*, 2024, **3**, 025302.
- 165 X. Wei, F. Yan, Q. Lv, W. Zhu, C. Hu, A. Patané and K. Wang, *Adv. Opt. Mater.*, 2019, **7**, 1900190.
- 166 J. Lu, Z. Zheng, J. Yao, W. Gao, Y. Xiao, M. Zhang and J. Li, *Nanoscale*, 2020, **12**, 7196–7205.
- 167 J. Chen, Z. Zhang, J. Feng, X. Xie, A. Jian, Y. Li, H. Guo, Y. Zhu, Z. Li, J. Dong, Q. Cui, Z. Shi and C. Xu, *Adv. Mater. Interfaces*, 2022, **9**, 2200075.
- 168 C. Liu, T. Zheng, K. Shu, S. Shu, Z. Lan, M. Yang, Z. Zheng, N. Huo, W. Gao and J. Li, *ACS Appl. Mater. Interfaces*, 2024, **16**, 13914–13926.
- 169 H. Wang, H. Huang, J. Zha, Y. Xia, P. Yang, Y. Zeng, Y. Liu, R. Cao, B. Wang, W. Wang, L. Zheng, Y. Chen, Q. He, X. Chen, K. Jiang, J. H. Lin, Z. Shi, J. C. Ho, H. Zhang and C. Tan, *Adv. Opt. Mater.*, 2023, **11**, 2301508.
- 170 H. Ghanbari, A. Abnavi, R. Ahmadi, M. R. Mohammadzadeh, M. Fawzy, A. Hasani and M. M. Adachi, *Adv. Opt. Mater.*, 2024, 2401682.
- 171 V. A. Semkin, A. V. Shabanov, D. A. Mylnikov, M. A. Kashchenko, I. K. Domaratskiy, S. S. Zhukov and D. A. Svintsov, *Nano Lett.*, 2023, **23**, 5250–5256.
- 172 V. Semkin, D. Mylnikov, E. Titova, S. Zhukov and D. Svintsov, *Appl. Phys. Lett.*, 2022, **120**, 191107.
- 173 X. Lu, L. Sun, P. Jiang and X. Bao, *Adv. Mater.*, 2019, **31**, 1902044.
- 174 G. Li, S. Yin, C. Tan, L. Chen, M. Yu, L. Li and F. Yan, *Adv. Funct. Mater.*, 2021, **31**, 2104787.
- 175 C. Xia, T. Yu, J. Liu, X. Li, J. Wang, S. Ma, G. Yang, H. Qiu, Z. Hu, N. Ye, J. Wang and Y. Wu, *Adv. Opt. Mater.*, 2024, **12**, 2400314.
- 176 Y. Niu, Y. Wang, W. Wu, P. Wang, J. Sun, J. Wen, X. Lu, P. Jiang, S. Zhang, N. Wang, D. Wu and Z. Zhao, *Adv. Opt. Mater.*, 2020, **8**, 2000833.
- 177 G. Li, H. Zhang, Y. Li, S. Yin, X. Kan, W. Wei, H. Du, B. Ge, C. An and M. Tian, *Nano Res.*, 2022, **15**, 5469–5475.
- 178 M. Dai, C. Wang, M. Ye, S. Zhu, S. Han, F. Sun, W. Chen, Y. Jin, Y. Chua and Q. J. Wang, *ACS Nano*, 2022, **16**, 295–305.
- 179 S. Lee, K. Hippalgaonkar, F. Yang, J. Hong, C. Ko, J. Suh, K. Liu, K. Wang, J. J. Urban, X. Zhang, C. Dames, S. A. Hartnoll, O. Delaire and J. Wu, *Science*, 2017, **355**, 371–374.
- 180 C. Liu, Y. Liu, Z. Chen, S. Zhang, C. Shi, G. Li, X. Yu, Z. Xu, L. Zhang, W. Zhao, X. Chen, W. Lu and L. Wang, *npj 2D Mater. Appl.*, 2022, **6**, 26.
- 181 W. Guo, Z. Dong, Y. Xu, C. Liu, D. Wei, L. Zhang, X. Shi, C. Guo, H. Xu, G. Chen, L. Wang, K. Zhang, X. Chen and W. Lu, *Adv. Sci.*, 2020, **7**, 1902699.
- 182 J. Zhang, X. Feng, Y. Xu, M. Guo, Z. Zhang, Y. Ou, Y. Feng, K. Li, H. Zhang, L. Wang, X. Chen, Z. Gan, S. C. Zhang, K. He, X. Ma, Q. K. Xue and Y. Wang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 075431.
- 183 X. Cai, A. B. Sushkov, R. J. Suess, M. M. Jadidi, G. S. Jenkins, L. O. Nyakiti, R. L. Myers-Ward, S. Li, J. Yan, D. K. Gaskill, T. E. Murphy, H. D. Drew and M. S. Fuhrer, *Nat. Nanotechnol.*, 2014, **9**, 814–819.
- 184 M. Razeghi, J. Spiece, O. Oğuz, D. Pehlivanoglu, Y. Huang, A. Sheraz, U. Başçı, P. S. Dobson, J. M. R. Weaver, P. Gehring and T. S. Kasirga, *npj 2D Mater. Appl.*, 2023, **7**, 36.
- 185 D. Li, J. K. Qin, B. Zhu, L. Q. Yue, P. Y. Huang, C. Zhu, F. Zhou, L. Zhen and C. Y. Xu, *ACS Nano*, 2024, **18**, 9636–9644.
- 186 Y. Li, J. Fu, X. Mao, C. Chen, H. Liu, M. Gong and H. Zeng, *Nat. Commun.*, 2021, **12**, 5896.
- 187 Y. R. Chang, R. Nanae, S. Kitamura, T. Nishimura, H. Wang, Y. Xiang, K. Shinokita, K. Matsuda, T. Taniguchi, K. Watanabe and K. Nagashio, *Adv. Mater.*, 2023, **35**, 2301172.
- 188 R. Nanae, S. Kitamura, Y. R. Chang, K. Kanahashi, T. Nishimura, R. Moqbel, K. H. Lin, M. Maruyama, Y. Gao, S. Okada, K. Qi, J. H. Fu, V. Tung, T. Taniguchi, K. Watanabe and K. Nagashio, *Adv. Funct. Mater.*, 2024, 2406140.
- 189 J. Wang, N. Han, Z. Lin, S. Hu, R. Tian, M. Zhang, Y. Zhang, J. Zhao and X. Gan, *Nanoscale*, 2024, **16**, 3101–3106.
- 190 V. K. Sangwan, D. G. Chica, T. C. Chu, M. Cheng, M. A. Quintero, S. Hao, C. E. Mead, H. Choi, R. Zu, J. Sheoran, J. He, Y. Liu, E. Qian, C. C. Laing, M. A. Kang, V. Gopalan, C. Wolverton, V. P. Dravid, L. J. Lauhon, M. C. Hersam and M. G. Kanatzidis, *Sci. Adv.*, 2024, **10**, eado8272.
- 191 N. Urakami, S. Ozaki and Y. Hashimoto, *Appl. Phys. Lett.*, 2024, **125**, 073102.
- 192 S. M. Nahid, S. Nam and A. M. van der Zande, *ACS Nano*, 2024, **18**, 14198–14206.
- 193 G. Xue, Z. Zhou, Q. Guo, Y. Zuo, W. Wei, J. Yang, P. Yin, S. Zhang, D. Zhong, Y. You, X. Sui, C. Liu, M. Wu, H. Hong, Z. J. Wang, P. Gao, Q. Li, L. Zhang, D. Yu, F. Ding, Z. Wei, C. Liu and K. Liu, *Science*, 2024, **384**, 1100–1104.
- 194 J. Wei, Y. Li, L. Wang, W. Liao, B. Dong, C. Xu, C. Zhu, K. W. Ang, C. W. Qiu and C. Lee, *Nat. Commun.*, 2020, **11**, 6404.
- 195 J. Xie, Z. Ren, J. Wei, W. Liu, J. Zhou and C. Lee, *Adv. Opt. Mater.*, 2023, **11**, 2202867.

- 196 S. M. Koepfli, M. Baumann, R. Gadola, S. Nashashibi, Y. Koyaz, D. Rieben, A. C. Güngör, M. Doderer, K. Keller, Y. Fedoryshyn and J. Leuthold, *Nat. Commun.*, 2024, **15**, 7351.
- 197 J. Zhang, M. Cai, X. Zheng, D. Li, S. Ma, X. Li, J. Fu, Y. Yuan, L. Chen, X. Guo, Y. Zhu and S. Zhuang, *APL Photonics*, 2024, **9**, 066105.
- 198 L. Zhang, X. Wang, Z. Zheng, C. Zhang, H. Zheng, C. Liu, H. Chen and M. Wang, *Nano Lett.*, 2024, **24**, 11654–11660.
- 199 Y. Hu, B. Y. Zhang, F. Haque, G. Ren and J. Z. Ou, *Mater. Horiz.*, 2022, **9**, 2288–2324.
- 200 C. Guo, Y. Hu, G. Chen, D. Wei, L. Zhang, Z. Chen, W. Guo, H. Xu, C. N. Kuo and C. S. Lue, *Sci. Adv.*, 2020, **6**, eabb6500.
- 201 W. Tang, A. Politano, C. Guo, W. Guo, C. Liu, L. Wang, X. Chen and W. Lu, *Adv. Funct. Mater.*, 2018, **28**, 1801786.
- 202 J. Chen and X. Ying, *AIP Adv.*, 2024, **14**, 035025.
- 203 S. Guo, Y. He, X. Lv, M. Jiang, Y. Wei, Y. Deng, X. Pan, S. Lan, D. Wang, A. Liu, C. Guo and L. Wang, *2D Mater.*, 2024, **11**, 035008.
- 204 C. M. Went, J. Wong, P. R. Jahelka, M. Kelzenberg, S. Biswas, M. S. Hunt, A. Carbone and H. A. Atwater, *Sci. Adv.*, 2019, **5**, eaax6061.
- 205 M. I. B. Utama, H. Kleemann, W. Zhao, C. S. Ong, F. H. da Jornada, D. Y. Qiu, H. Cai, H. Li, R. Kou, S. Zhao, S. Wang, K. Watanabe, T. Taniguchi, S. Tongay, A. Zettl, S. G. Louie and F. Wang, *Nat. Electron.*, 2019, **2**, 60–65.
- 206 Y. Xu, C. Horn, J. Zhu, Y. Tang, L. Ma, L. Li, S. Liu, K. Watanabe, T. Taniguchi, J. C. Hone, J. Shan and K. F. Mak, *Nat. Mater.*, 2021, **20**, 645–649.
- 207 H. Ma, H. Fang, J. Li, Z. Li, X. Fang and H. Wang, *InfoMat*, 2024, **6**, e12540.
- 208 X. Tang, S. Wang, Y. Liang, D. Bai, J. Xu, Y. Wang, C. Chen, X. Liu, S. Wu, Y. Wen, D. Jiang and Z. Zhang, *Phys. Chem. Chem. Phys.*, 2022, **24**, 7323–7330.
- 209 B. Qin, C. Ma, Q. Guo, X. Li, W. Wei, C. Ma, Q. Wang, F. Liu, M. Zhao, G. Xue, J. Qi, M. Wu, H. Hong, L. Du, Q. Zhao, P. Gao, X. Wang, E. Wang, G. Zhang, C. Liu and K. Liu, *Science*, 2024, **385**, 99–104.
- 210 L. Li, Q. Wang, F. Wu, Q. Xu, J. Tian, Z. Huang, Q. Wang, X. Zhao, Q. Zhang, Q. Fan, X. Li, Y. Peng, Y. Zhang, K. Ji, A. Zhi, H. Sun, M. Zhu, J. Zhu, N. Lu, Y. Lu, S. Wang, X. Bai, Y. Xu, W. Yang, N. Li, D. Shi, L. Xian, K. Liu, L. Du and G. Zhang, *Nat. Commun.*, 2024, **15**, 1825.
- 211 J. Jiang, L. Xu, L. Du, L. Li, G. Zhang, C. Qiu and L. M. Peng, *Nat. Electron.*, 2024, **7**, 545–556.
- 212 C. Li, F. Zheng, J. Min, N. Yang, Y. M. Chang, H. Liu, Y. Zhang, P. Yang, Q. Yu, Y. Li, Z. Luo, A. Aljarb, K. Shih, J. K. Huang, L. J. Li and Y. Wan, *Adv. Mater.*, 2024, 2404923.
- 213 Q. Yang, X. Wang, Z. He, Y. Chen, S. Li, H. Chen and S. Wu, *Adv. Sci.*, 2023, **10**, 2205609.
- 214 O. Maßmeyer, R. Günkler, J. Glowatzki, P. Klement, B. Ojaghi Dogahe, S. R. Kachel, F. Gruber, M. Müller, M. Fey, J. Schörmann, J. Belz, A. Beyer, J. M. Gottfried, S. Chatterjee and K. Volz, *Small*, 2024, **20**, 2402155.
- 215 S. Kim, W. Lee, K. Ko, H. Cho, H. Cho, S. Jeon, C. Jeong, S. Kim, F. Ding and J. Suh, *Adv. Mater.*, 2024, **36**, 2400800.
- 216 Y. Wang, C. Zhao, X. Gao, L. Zheng, J. Qian, X. Gao, J. Li, J. Tang, C. Tan, J. Wang, X. Zhu, J. Guo, Z. Liu, F. Ding and H. Peng, *Nat. Mater.*, 2024, DOI: [10.1038/s41563-01024-01968-z](https://doi.org/10.1038/s41563-01024-01968-z).
- 217 A. Rajan, S. Buchberger, B. Edwards, A. Zivanovic, N. Kushwaha, C. Bigi, Y. Nanao, B. K. Saika, O. R. Armitage, P. Wahl, P. Couture and P. D. C. King, *Adv. Mater.*, 2024, **36**, 2402254.
- 218 N. Chapuis, A. Mahmoudi, C. Coinon, D. Troadec, D. Vignaud, G. Patriarche, P. Roussel, A. Ouerghi, F. Oehler and X. Wallart, *2D Mater.*, 2024, **11**, 035031.
- 219 Y. Ma, Z. Deng, H. Liang, X. Guan, Z. Zheng, J. Yao and G. Yang, *Laser Photonics Rev.*, 2024, DOI: [10.1002/lpor.202400669](https://doi.org/10.1002/lpor.202400669).
- 220 K. Lee, N. Higashitarumizu, S. Wang, C. Kim, C. Y. Ho, J. W. Oh, G. Zan, M. Madsen, T. W. Lee, D. C. Chrzan, C. Park and A. Javey, *Sci. Adv.*, 2024, **10**, eadq0655.
- 221 M. Kim, Y. Lee, K. Kim, G. H. Pham, K. Kim, J. H. Jun, H. Lee, S. Yoon, H. J. Hwang, M. M. Sung and B. H. Lee, *Nanoscale Horiz.*, 2024, DOI: [10.1039/D1034NH00339J](https://doi.org/10.1039/D1034NH00339J).
- 222 R. Wadhwa, S. Thapa, S. Deswal, P. Kumar and M. Kumar, *J. Phys.: Condens. Matter*, 2023, **35**, 124002.
- 223 F. Xia, D. Wang, J. Cao, Z. Zeng, B. Zhang, J. Pan, D. Liu, S. Liu, C. Zhao, S. Jiao, T. Chen, G. Liu, X. Fang, L. Zhao and J. Wang, *Sci. China Mater.*, 2024, **67**, 2293–2301.