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Hierarchical vanadium sulfide nanosheets with expanded interchain spacing for high-performance sodium-ion batteries

Jianbiao Wang,^{†*a} Peidian Chong,^{†b} Wanwisa Limphirat,^{†c} Haiyi Wang,^a Wutthikrai Busayaporn,^c Lei Zhang,^a Debbie Hwee Leng Seng,^a Shengnan Sun,^a Zainul Aabdin,^a Chaoyu Dong,^d Mingdeng Wei^{*b} and Zhi Wei Seh^{*a}

Sodium-ion batteries (SIBs) have been intensively researched as potential alternative energy storage devices to lithium-ion batteries (LIBs). Nevertheless, the scarcity of suitable anode materials capable of hosting the large radius of Na⁺ has hindered the further application of SIBs. Herein, we developed a hierarchical VS₄ nanosheet with an expanded interchain spacing of 0.98 nm without additives for the first time. Additionally, we found that the porous structure in the hierarchical VS₄ nanosheet provides sufficient active sites for Na⁺ storage and alleviates the volume variation during discharge/charge cycles, as supported by finite element simulation (FES) data. More importantly, a dynamic insertion-dominated storage mechanism was revealed through synchrotron X-ray absorption spectroscopy and X-ray photoelectron spectroscopy. Thus, the optimized anode delivered a high capacity of 441 mAh g⁻¹ at 1 A g⁻¹ after 200 cycles. This work provides critical insights into the design of SIBs by correlating storage mechanisms with electrode's structural composition.

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1. Introduction

With the mass application of lithium-ion batteries (LIBs), increasing concerns over cost and safety have prompted research interest in alternative energy storage devices, including sodium-ion batteries and multivalent-ion batteries.^{1–5} Sodium-ion batteries (SIBs) are considered as a promising alternative, due to their similar redox chemistry to LIBs, as well as the natural abundance of sodium.^{6–10} Nevertheless, the larger radius of Na⁺ than that of Li⁺ leads to sluggish electrochemical kinetics, hampering the practical implementation of SIBs.^{11–17}

The electrochemical kinetics of SIBs can be enhanced by employing transition metal sulfides as anodes, owing to their high theoretical capacity and excellent ionic conductivity.^{7,18–20} Among these, VS₄ has emerged as a promising candidate, as its abundant (S₂)^{2–} anions contribute to high capacity. However, its practical application is hindered by significant volume changes during the discharge/charge process.^{21–24} To address this challenge, various strategies have been explored, such as combining VS₄ with MXenes or fabricating heterojunction



Jianbiao Wang

*Dr Jianbiao Wang is a Scientist at the Institute of Materials Research and Engineering (IMRE), Agency for Science, Technology and Research (A*STAR), Singapore. He received his PhD in 2021 from Fuzhou University. From 2019 to 2021, he served as a Special Researcher at Nagasaki University, Japan. His research interests include the development of novel electrode materials, the investigation of energy storage mechanisms*

involving Li⁺, Na⁺, Mg²⁺, and Zn²⁺ ion batteries, and the optimization of electrolyte systems.

^aInstitute of Materials Research and Engineering (IMRE), Agency for Science, Technology and Research (A*STAR), 2 Fusionopolis Way, Innovis #08-03, Singapore 138634, Republic of Singapore. E-mail: jianbiao_wang@imre.a-star.edu.sg; sehzw@imre.a-star.edu.sg

^bFujian Key Laboratory of Electrochemical Energy Storage Materials, Fuzhou University, Fuzhou 350116, China. E-mail: wei-mingdeng@fzu.edu.cn

^cSynchrotron Light Research Institute, 111 Moo 6 University Avenue, Muang, Nakhon Ratchasima 30000, Thailand

^dSingapore Institute of Manufacturing Technology (SIMTech), Agency for Science, Technology and Research (A*STAR), 5 Cleantech Loop, 01-01 Cleantech Two Block B, Singapore 636732, Republic of Singapore

[†] These authors contributed equally to this work.

structures.²⁵ However, achieving high capacity VS₄-based anodes remains challenging.

Ether-based electrolytes with sodium trifluoromethanesulfonate (NaOTF) have been extensively researched in SIBs due to high reversibility during plating/stripping of Na⁺.^{6,9} A limited understanding of the storage mechanisms of VS₄ in NaOTF-ether electrolytes also further constrains its application in SIBs.

Here, for the first time, we synthesized hierarchical VS₄ nanosheets with expanded interchain spacings *via* a one-step solvothermal process, without the need for any additives. To the best of our knowledge, this work reports the largest interchain spacing for VS₄ to date, offering increased Na⁺ storage capacity. Unlike previous studies, we unveiled a dynamic insertion-dominated storage mechanism in the discharge/charge process with NaOTF-ether electrolyte, using synchrotron X-ray absorption spectroscopy (XAS) and X-ray photoelectron spectroscopy (XPS). Additionally, the porous structure provides ample space to accommodate Na⁺ and mitigates the volume variation in the insertion/extraction of Na⁺, enhancing structural stability, as supported by finite element simulation (FES) data. The presence of nanosheets can effectively reduce the diffusion pathway of Na⁺. Moreover, the composition of the solid electrolyte interface (SEI) formed on the surface of VS₄ nanosheets was also investigated using XPS. Thus, a stable electrochemical performance of VS₄ could be achieved. In detail, the hierarchical VS₄ nanosheet anode retains a high capacity of 441 mAh g⁻¹ at 1000 mA g⁻¹ after 200 cycles, showing advantages among VS₄-based anodes.¹⁹ Based on this work, we found that the interfacial chemistry between the electrode and electrolyte is vital for the cycling performance of batteries. Future efforts will focus on investigating and optimizing SEI formation to advance the development of high-energy-density batteries.

2. Experimental

2.1 Synthesis of hierarchical nanosheets

For the synthesis of hierarchical VS₄ nanosheets, 1 mmol of NH₄VO₃ was added into 30 ml of ethanol and stirred for about 5 min. Then, 4 mmol of thioacetamide was added to the above solution and stirred for another 10 min. Subsequently, the solution was transferred to a 200 °C oven and heated for 12 h. After cooling naturally to room temperature, the black product was collected *via* centrifugation and washed with ethanol and isopropanol several times. At last, the product was obtained by drying in a vacuum oven at 70 °C overnight.

2.2 Electrochemical measurement

Detailed information about electrochemical measurements is provided in the SI.

2.3 Material characterization

Please see the information regarding material characterization in the SI.

2.4 FES method

Finite element simulations were performed using COMSOL software. A hygroscopic model was employed to represent the swelling behaviour resulting from the absorption of Na⁺ ions, using a concentration of 1 mol L⁻¹. The hygroscopic swelling coefficient was assumed to be 1 × 10⁻³ m³ kg⁻¹. As the parameters may not accurately represent real conditions, the results are intended for qualitative analysis only. To simulate the displacement distribution caused by swelling, the external surfaces of the nanoparticles were set as free boundaries.

3. Results and discussion

Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images are presented in Fig. 1 to investigate the morphological and structural composition of VS₄. A hierarchical VS₄ nanosheet with a porous structure is observed, where the magnified SEM images reveal that the porous structure consists of interconnected nanosheets (Fig. 1a and b). The TEM image in Fig. 1c reveals the characteristics of nanosheets. The high-resolution TEM (HRTEM) image shows lattice fringes of 0.98 nm, indicating an expanded interchain spacing along the (110) plane, in contrast with the conventional spacing of 0.58 nm in previous reports (Fig. 1d).^{19,20} Moreover, the expanded interchain spacing is also evident from lattice fringes at the edges, which could facilitate enhanced Na⁺ insertion/extraction. We suppose that the expanded interchain spacing could be ascribed to the production of gas from NH₄VO₃ in the synthetic process, which results in the interchain spacing expansion to form hierarchical nanosheets. Scanning transmission electron microscopy (STEM) and corresponding elemental maps reveal the even distribution of V and S around the hierarchical nanosheets (Fig. 1e and f).

As presented in Fig. 2a, most reflections in the XRD pattern are well indexed to standard reference of VS₄. Notably, the characteristic peak of the (110) plane is observed to be shifted to 8.8°, and the corresponding interchain spacing is calculated to be 0.99 nm based on the Braggs law equation.¹⁶ The calculated expanded interchain spacing is greater than previous reported spacing, agreeing well with TEM observation.^{19,20} This is also the first report of such a large interchain spacing in VS₄, which could allow for more Na⁺ to insert into the structures. The observed bands at 139 cm⁻¹, 191 cm⁻¹, 281 cm⁻¹, and 405 cm⁻¹ from the Raman spectrum are ascribed to characteristics of VS₄ (Fig. 2b).^{26,27} The survey X-ray photoelectron spectra confirmed the presence of V and S elements in VS₄ (Fig. S1). The curve-fitted peak components of V 2p at 524.2 eV and 516.8 eV, as well as 521.2 eV and 513.7 eV, are associated with V⁴⁺ and V³⁺, respectively, which agrees well with previous reports.^{28–30} Curve fitting of the S 2p spectrum yields three peaks: the binding energies at 163.5 eV and 162.3 eV are assigned to S 2p_{1/2} and S 2p_{3/2} of (S₂)²⁻, respectively, while the binding energy at 161.1 eV is related to S²⁻.^{21,22,27} These characterization results confirmed the successful preparation of hierarchical VS₄ nanosheets.

To gain deeper insight into the formation mechanism of hierarchical VS₄ nanosheets, we have conducted experiments



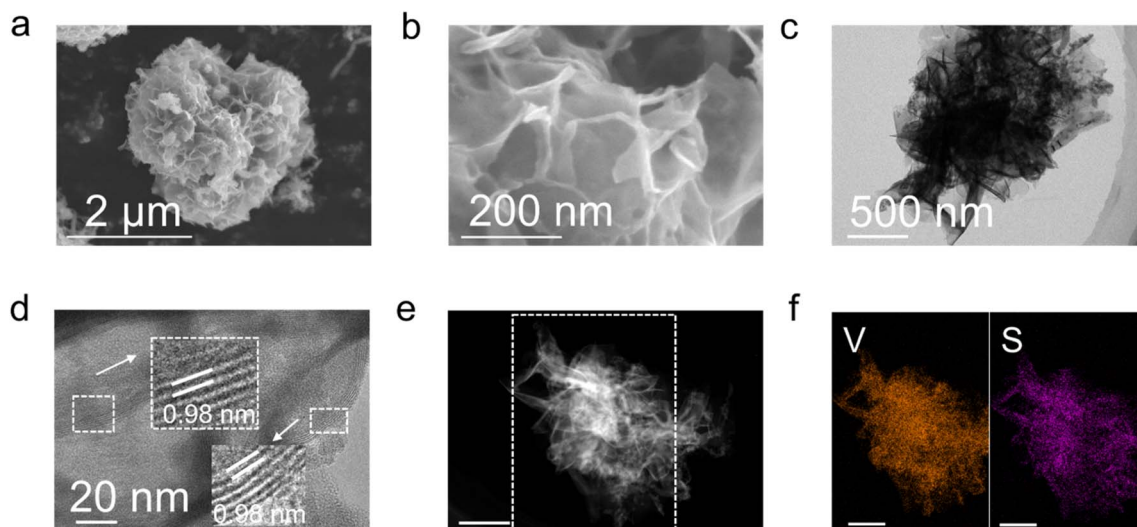


Fig. 1 Material characterization: (a and b) SEM images, (c) TEM image, (d) HRTEM image, and (e) STEM image and (f) corresponding elemental maps of V and S in hierarchical VS_4 nanosheets (scale bar: (e): 500 nm, (f): 500 nm).

with different reaction times of 10 min, 30 min, and 1 h. Then we collected the TEM images and corresponding maps of these samples, as demonstrated in Fig. S2–4. Based on these observations, we propose the formation mechanism of hierarchical VS_4 nanosheets, as demonstrated in Fig. S5. The evolution of

crystallinity is revealed by XRD patterns with different reaction times (Fig. S6). The characteristic reflections of VS_4 appeared when the reaction time reached up to 2 h.

We conducted a series of electrochemical tests to evaluate the effectiveness of the designed hierarchical VS_4 nanosheets in

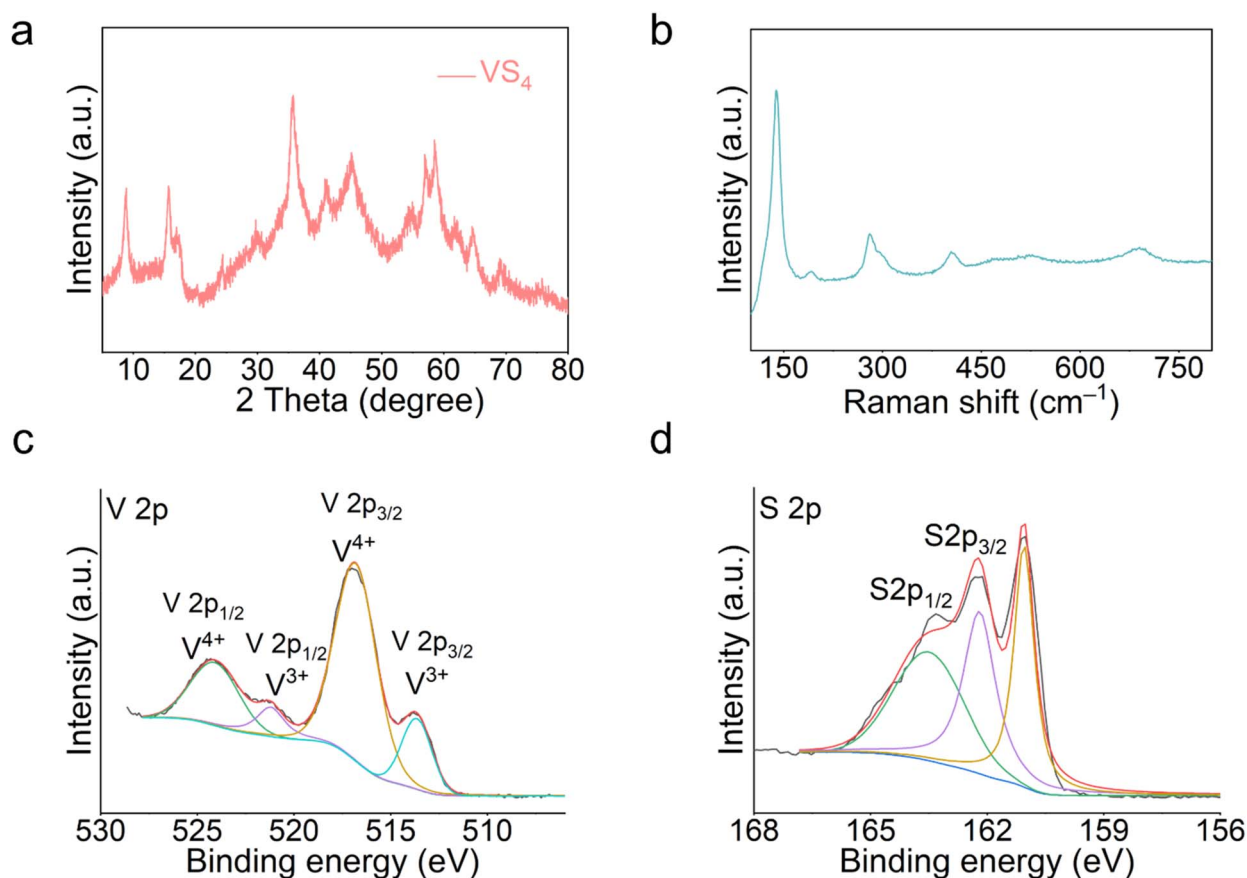


Fig. 2 Materials characterization: (a) XRD pattern, (b) Raman spectrum, and XPS spectra of (c) V 2p and (d) S 2p, respectively.





Fig. 3 Electrochemical performances of hierarchical VS₄ nanosheets: (a) CV curves at 0.5 mV s⁻¹, cycling performance at current densities of (b) 500 mA g⁻¹ and (c) 1000 mA g⁻¹, and (d) the corresponding charge-discharge profiles after different cycles at 1000 mA g⁻¹. (e) CV curves at varying scan rates and (f) the corresponding calculated values of *b*.

hosting Na⁺. CV curves were obtained at 0.5 mV s⁻¹, in which the decreased voltage gap of redox peaks could be observed upon cycling (Fig. 3a). Moreover, the increasing intensity of redox peaks in the CV curves over successive cycles indicates reduced polarization and accelerated kinetics. Thereafter, the cycling performance was tested at 500 mA g⁻¹, and a stable

capacity of 509 mAh g⁻¹ is achieved after 110 cycles (Fig. 3b). Even at a higher current density of 1000 mA g⁻¹ (Fig. 3c), the electrode can retain a high capacity of 441 mAh g⁻¹ after 200 cycles, showing some advantages among reported VS₄-based anodes in SIBs (Table S1).^{21,31-41} Furthermore, the plateaus observed in corresponding charge-discharge profiles at 1000

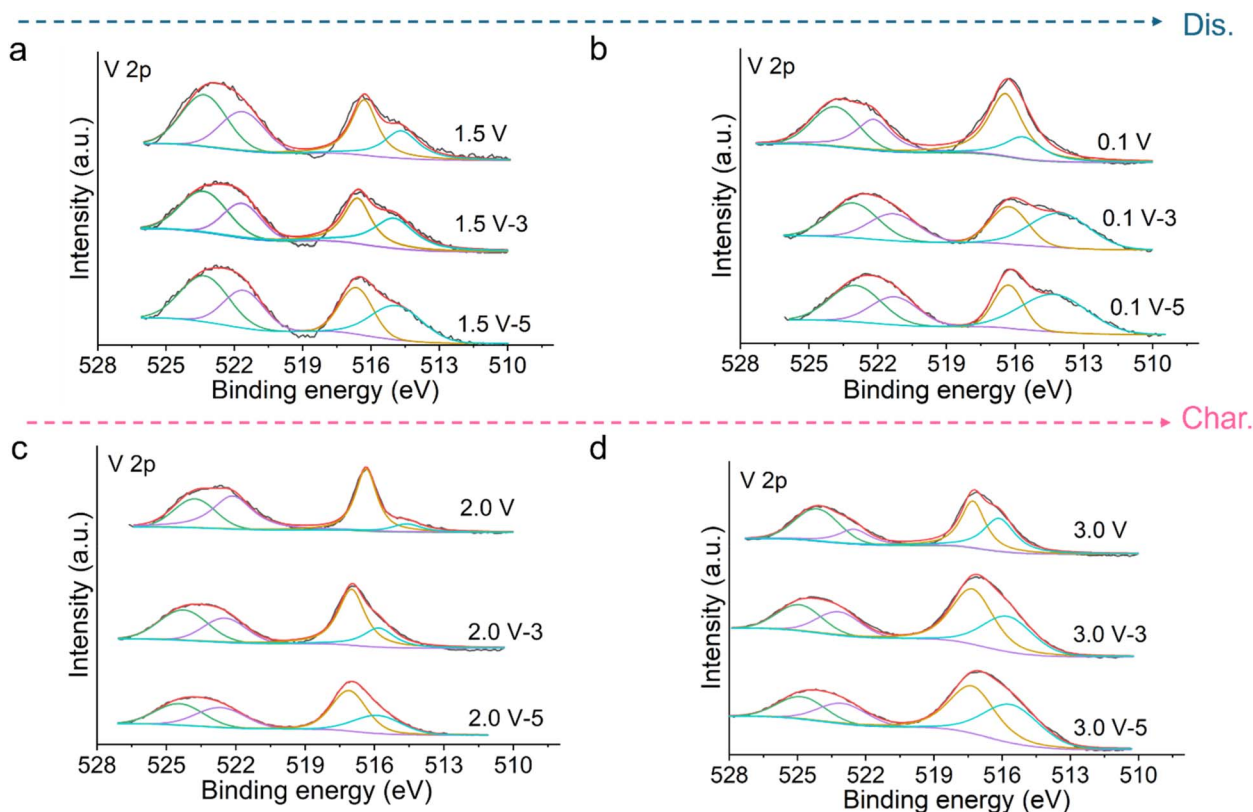


Fig. 4 *Ex situ* XPS spectra of V 2p from samples at different potential states of (a) 1.5 V, (b) 0.1 V, (c) 2.0 V, and (d) 3.0 V.



mA g^{-1} agree well with the redox peaks in CV curves. Moreover, the charge–discharge profiles after the 100th cycle demonstrate the smallest voltage gap between discharge and charge plateaus compared to the 10th and 50th cycles, further supporting the enhanced kinetics as observed in CV curves (Fig. 3d). In contrast, the electrode is short-circuited after 43 cycles with an inferior capacity of 75 mA h g^{-1} when it was used in electrolytes based on NaPF_6 , reflecting the advantages of NaOTf electrolytes (Fig. S7). CV measurements at varying scan rates were performed to investigate the capacity contribution dominated by the diffusion-controlled process or the capacitance effect, based on the following equation:

$$i = av^b \quad (1)$$

where i is the current, a and b are constants, and v is the sweep rate (Fig. 3e). The calculated b values represent whether the electrochemical process is dominated by the capacitance effect or the diffusion-controlled process, when the values of b are close to 1 or 0.5, respectively. The calculated values of b are 1.0 and 0.98 at anodic and cathodic peaks, respectively (Fig. 3f). These results indicate that the electrochemical process was dominated by the capacitance effect. The specific contribution from capacitance can be evaluated using the following equation:

$$i = k_1v + k_2v^{1/2} \quad (2)$$

in which i is the current, v is the sweep rate, and k_1 and k_2 are constants. The calculated results indicate that capacitive contribution increases with increasing sweep rates. In specific, the capacitive contribution approaches to 90.2% at a sweep rate of 1 mV s^{-1} (Fig. S8). To further demonstrate the benefits of the designed hierarchical VS_4 nanosheets, electrochemical impedance spectroscopy (EIS) shows a significantly lower impedance after 20 cycles than that after 3 cycles, indicating improved kinetics with ongoing cycling (Fig. S9).

XPS depth profiling and synchrotron XAS spectra are used to understand the storage mechanism and interfacial chemistry between hierarchical VS_4 nanosheets and electrolytes (Fig. 4–7). *Ex situ* XPS spectra were collected from samples at different potential states of 1.5 V, 0.01 V, 2.0 V, and 3.0 V. The data labelled 1.5 V, 1.5 V–3, and 1.5 V–5 represent samples treated with etching times of 0 min, 3 min, and 5 min, respectively (the notation applies to other potential states as well). Upon discharge to 1.5 V, the intensity peaks for V^{4+} increase, while those for V^{3+} decrease, in contrast with the pristine sample. This should be attributed to insertion of Na^+ and transfer of electrons within VS_4 and formation of Na_3VS_4 .²¹ No significant differences in V 2p are observed among the 1.5 V, 1.5 V–3, 1.5 V–5 samples. Upon further discharge to 0.1 V, the intensity ratio for $\text{V}^{4+}/\text{V}^{3+}$ increases at the electrode surface, while it decreases in both 0.1 V–3 and 0.1 V–5. This suggests the co-existence of insertion/conversion reactions during the discharge process,

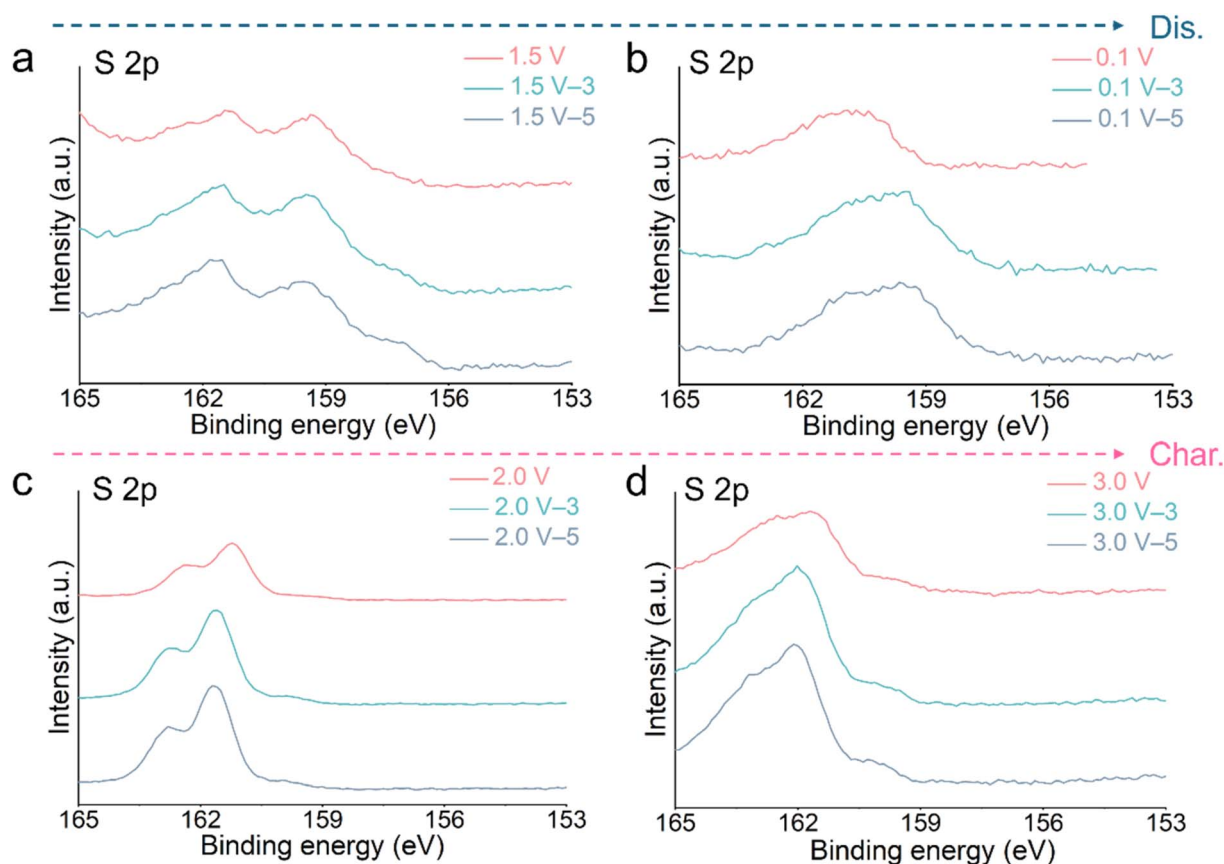


Fig. 5 *Ex situ* XPS spectra of S 2p collected at different voltages of (a) 1.5 V, (b) 0.1 V, (c) 2.0 V, and (d) 3.0 V, respectively.





Fig. 6 *Ex situ* XPS spectra of F 1s collected at different voltages of (a) 1.5 V, (b) 0.1 V, (c) 2.0 V, and (d) 3.0 V, respectively.

with the high V^{4+}/V^{3+} intensity ratio at 0.1 V likely attributed to the surface oxidation in the process of measurements.

During the subsequent charge process, the binding energies of V 2p at 2.0 V are higher than those at 1.5 V and 0.1 V, which comes from the presence of V^{5+} .²⁸ More importantly, the binding energies of V 2p at 2.0 V-3 and 2.0 V-5 are similar but higher than those at 2.0 V, suggesting that the degree of insertion reaction is greater within the bulk of the material than at

the surface. Upon full charging to 3.0 V, the V 2p binding energies shift further to higher values, demonstrating partial reversibility of the insertion reactions. These results suggest that Na_3VS_4 remains partially unconverted, with only partial reversibility of the conversion reactions. Therefore, the final product at the full charge state is likely a composite of Na_3VS_4 (V^{5+}) and VS_4 (V^{4+}), which is different from previous reports that use NaOTf-based ether electrolyte.^{21,30}

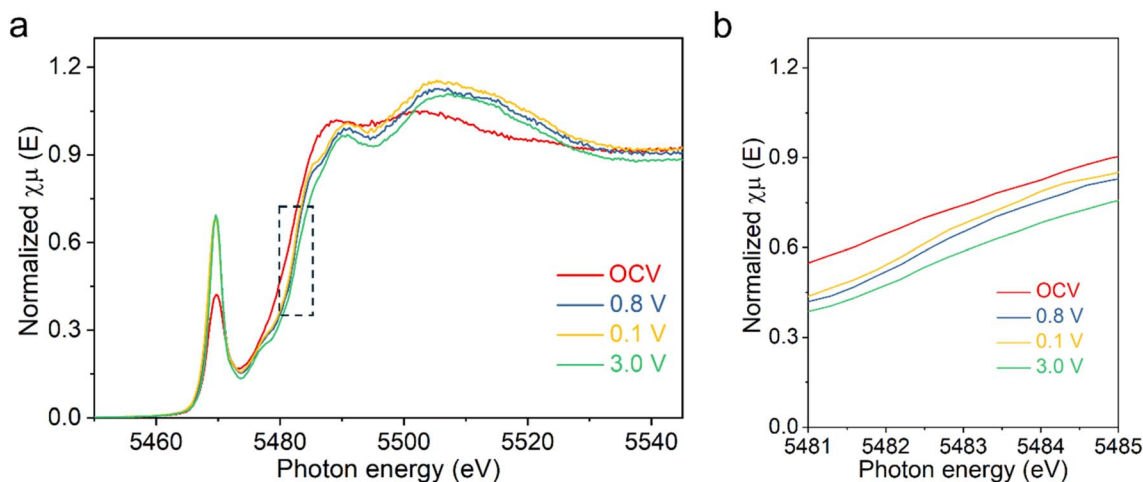


Fig. 7 (a) XAS spectra collected at different potential states, and (b) corresponding magnified part of the K-edge.





Fig. 8 FES: (a and c) stress distributions in solid spheres and hierarchical VS₄ nanosheets, respectively (b and d). Total displacements in solid spheres and hierarchical VS₄ nanosheets, respectively.

We have also investigated the composition of solid electrolyte interphase (SEI) formed on the surface of the anode after two cycles (Fig. 6 and S10). In detail, the deconvoluted F 1s spectra show binding energies at 688.5 eV and 683.7 eV, corresponding to Na–F and C–F bonds, respectively.^{11,42} The disappearance of the C–F peak after the etching treatment suggests that the C–F bonds originate from electrolytes. Moreover, the binding energy for Na–F binding can be observed at different etching times, indicating the formation of a stable NaF-rich SEI layer. The corresponding Na 1s XPS spectra collected from different potential states also supported the presence of a SEI formed on hierarchical VS₄ nanosheets, accounting for the superior electrochemical sodium storage performance.

XAS spectra at different potential states were collected to gain a deeper understanding of the energy storage mechanism (Fig. 7a and b). The K-edge of V shifts to higher photon energy in contrast to the sample at open circuit voltage (OCV) throughout the discharge/charge process, which is attributed to the insertion of Na⁺ and the formation of Na₃VS₄. Notably, at a discharge cut-off voltage of 0.1 V, the V K-edge approaches that of OCV, suggesting a hybrid mechanism involving both conversion and insertion. In the converse charge potential states of 0.8 V and 3.0 V, the V K-edges further shift to higher energy regions relative to 0.1 V and OCV. These observations confirm that the insertion reactions dominate the entire electrochemical process at the bulk level, aligning well with *ex situ* XPS results.

To highlight the advantages of hierarchical VS₄ nanosheets, we compared the stress and corresponding displacements

between solid spheres and hierarchical nanosheet structures of VS₄ under free boundary conditions (Fig. 8). The average stress in solid spheres and hierarchical nanosheets is simulated to be similar (Fig. 8a and c). However, the average displacement in solid spheres (22.8 nm) is higher than that in hierarchical nanosheets (19.8 nm) (Fig. 8b and d). Notably, displacement in the hierarchical nanosheets is mainly located at the outer edges, providing sufficient space to allow for volume change. In contrast, there is no space for solid spheres to undergo expansion, making them prone to cracking during cycling. So, these results demonstrate that the rational design of hierarchical nanosheets effectively mitigates mechanical stress and enhances electrochemical performance.

Computer tomography (CT) of pouch cells before cycling and after 20 cycles is used to investigate the volume changes of hierarchical VS₄ nanosheets (Fig. S11). The cross-section CT slices show no significant change in the anode thickness after 20 cycles, which is consistent with the FES results. Moreover, the TEM image and corresponding elemental maps of the electrode after 20 cycles reveal that the hierarchical nanosheet morphology can be well retained, supporting the structural stability of hierarchical VS₄ nanosheets (Fig. S12).

4. Conclusions

In this work, we have developed hierarchical VS₄ nanosheets with expanded interchain distance for the first time *via* a one-step solvothermal process without the use of additives. The



expanded interchain distance can accommodate more insertion/extraction of Na^+ , while the porous hierarchical nanosheets can alleviate the volume variation and reduce the diffusion pathway of Na^+ in the electrochemical process, as supported by data from finite element simulations. Furthermore, we uncovered an insertion behavior dominated sodium storage mechanism *via ex situ* XPS and XAS spectra. As such, the optimized anode demonstrates a high capacity of 441 mAh g^{-1} after 200 cycles at 1 A g^{-1} , highlighting the advantages of rationally designed hierarchical nanosheet structures. This work emphasizes the importance of engineering in both morphology and internal structure (e.g., interchain spacing) to achieve superior electrochemical performance. Moreover, this work also offers new insights into the working principles of VS_4 in SIBs. Future research will focus on further elucidating electrode storage mechanisms to advance the understanding and development of high-performance battery systems.

Author contributions

J. W., P. C., and W. L. contributed equally to this work. J. W. and Z. W. S. conceived the original concept and initiated the project. J. W. wrote the manuscript. M. W., H. W. and Z. W. S. revised it. P. C. synthesized the materials and performed the electrochemical performance test. Z. A. performed TEM, S. S. carried out XRD, W. L. and W. B. conducted XAS, and L. Z. carried out the simulations.

Conflicts of interest

There are no conflicts to declare.

Data availability

The data supporting this article has been included as part of the SI.

Supplementary information is available. See DOI: <https://doi.org/10.1039/d5ta03608a>.

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References

- 1 Y. Liang and Y. Yao, *Nat. Rev. Mater.*, 2022, **8**, 109–122.
- 2 A. Y. S. Eng, C. B. Soni, Y. Lum, E. Khoo, Z. Yao, S. Vineeth, V. Kumar, J. Lu, C. S. Johnson and C. Wolverton, *Sci. Adv.*, 2022, **8**, eabm2422.
- 3 J. Wang, G. Yang, T. Ghosh, Y. Bai, C. Y. J. Lim, L. Zhang, D. H. L. Seng, W. P. Goh, Z. Xing, Z. Liu and Z. W. Seh, *Nano Energy*, 2024, **119**, 109082.
- 4 J. Wang, T. Ghosh, Z. Ju, M.-F. Ng, G. Wu, G. Yang, X. Zhang, L. Zhang, A. D. Handoko and S. Kumar, *Matter*, 2024, **7**, 1833–1847.
- 5 J. Wang and Z. W. Seh, *Acc. Mater. Res.*, 2024, **5**, 1329–1339.
- 6 J. Wang, N. Luo, J. Wu, S. Huang, L. Yu and M. Wei, *J. Mater. Chem. A*, 2019, **7**, 3691–3696.
- 7 J. Wang, L. Han, X. Li, L. Zeng and M. Wei, *J. Colloid Interf. Sci.*, 2019, **548**, 20–24.
- 8 Y. Gao, Z. Wang, H. Tu, J. Xue, S. Weng, S. Lu, L. Liu, G. Sun, K. Peng, X. Zhang, D. Li, Y. Liu, J. Xu, H. Li and X. Wu, *Adv. Funct. Mater.*, 2024, **35**.
- 9 C. Liu, K. Chen, F. Li, A. Zhao, P. Liu, Z. Chen, Y. Fang and Y. Cao, *J. Am. Chem. Soc.*, 2025, **147**, 14635–14646.
- 10 H. Chen, K. Chen, J. Yang, B. Liu, L. Luo, H. Li, L. Chen, A. Zhao, X. Liang, J. Feng, Y. Fang and Y. Cao, *J. Am. Chem. Soc.*, 2024, **146**, 15751–15760.
- 11 Z. W. Seh, J. Sun, Y. Sun and Y. Cui, *ACS Cent. Sci.*, 2015, **1**, 449–455.
- 12 X. Lu, H. Zhao, Y. Qin, E. Matios, J. Luo, R. Chen, H. Nan, B. Wen, Y. Zhang, Y. Li, Q. He, X. Deng, J. Lin, K. Zhang, H. Wang, K. Xi, Y. Su, X. Hu, S. Ding and W. Li, *ACS Nano*, 2023, **17**, 10665–10676.
- 13 C. Wang, A. C. Thenuwara, J. Luo, P. P. Shetty, M. T. McDowell, H. Zhu, S. Posada-Perez, H. Xiong, G. Hautier and W. Li, *Nat. Commun.*, 2022, **13**, 4934.
- 14 A. Rudola, R. Sayers, C. J. Wright and J. Barker, *Nat. Energy*, 2023, **8**, 215–218.
- 15 C. Vaalma, D. Buchholz, M. Weil and S. Passerini, *Nat. Rev. Mater.*, 2018, **3**, 1–11.
- 16 F. Zhang, B. He, Y. Xin, T. Zhu, Y. Zhang, S. Wang, W. Li, Y. Yang and H. Tian, *Chem. Rev.*, 2024, **124**, 4778–4821.
- 17 Y. Zhao, Y. Kang, J. Wozny, J. Lu, H. Du, C. Li, T. Li, F. Kang, N. Tavajohi and B. Li, *Nat. Rev. Mater.*, 2023, **8**, 623–634.
- 18 J. Wang, L. Yu, Z. Zhou, L. Zeng and M. Wei, *J. Colloid Interf. Sci.*, 2019, **557**, 722–728.
- 19 J. Wang, J. Huang, S. Huang, H. Notohara, K. Urita, I. Moriguchi and M. Wei, *ACS Sustain. Chem. Eng.*, 2020, **8**, 9519–9525.
- 20 J. Wang, J. Okabe, K. Urita, I. Moriguchi and M. Wei, *J. Electroanal. Chem.*, 2020, **874**, 114523.
- 21 S. Wang, F. Gong, S. Yang, J. Liao, M. Wu, Z. Xu, C. Chen, X. Yang, F. Zhao, B. Wang, Y. Wang and X. Sun, *Adv. Funct. Mater.*, 2018, **28**, 1801806.
- 22 L. q. Yu, S. X. Zhao, Q. l. Wu, J. W. Zhao and G. d. Wei, *Adv. Funct. Mater.*, 2020, **30**, 2000427.
- 23 Z. Wang, X. Li, W. Guo and Y. Fu, *Adv. Funct. Mater.*, 2021, **31**, 2009875.
- 24 D. Zhang, Y. Shao, J. Wang, Z. Li, Q. Wang, H. Sun, Q. Sun and B. Wang, *Small Struct.*, 2023, **5**, 2300217.
- 25 M. Tang, X. Yin, Y. Cao, H. Ma, X. Zhang and D. Jia, *Small*, 2024, **20**, e2406547.
- 26 D. Zhang, Y. Shao, J. Wang, Z. Li, Q. Wang, H. Sun, Q. Sun and B. Wang, *Small*, 2024, **20**, e2309901.
- 27 D. Yang, S. Zhang, P. Yu, S. Cheng, Z. Yuan, Y. Jiang, W. Sun, H. Pan, Y. Feng, X. Rui and Y. Yu, *Small*, 2022, **18**, e2107058.
- 28 P. Yu, S. Xu, K. Yao, H. Yao, W. Yang, X. Lin, H. Yu, W. Liu, Y. Qin and X. Rui, *J. Power Sources*, 2021, **501**, 230021.



- 29 W. Li, J. Huang, R. Li, L. Cao, X. Li, L. Feng and S. Chen, *Chem. Eng. J.*, 2020, **384**, 123385.
- 30 Z. Qin, Y. Hu, C. Lv, S. Yao and G. Chen, *Chem. Eng. J.*, 2022, **433**, 133765.
- 31 L. Song, Y. Tang, R. Mao, J. Sun, C. Yu, Y. Liu and Y. Zhao, *ACS Appl. Nano Mater.*, 2024, **7**, 9002–9011.
- 32 W. Li, J. Huang, R. Li, L. Cao, X. Li, S. Chen and L. Feng, *ChemSusChem*, 2019, **12**, 5183–5191.
- 33 X. Li, H. Liang, B. Qin, M. Wang, Y. Zhang and H. Fan, *J. Colloid Interf. Sci.*, 2022, **625**, 41–49.
- 34 F. Yang, W. Zhong, H. Wang, M. Ren, W. Liu, M. Li and L. Su, *J. Alloys and Compd.*, 2020, **834**, 155204.
- 35 R. Sun, Q. Wei, Q. Li, W. Luo, Q. An, J. Sheng, D. Wang, W. Chen and L. Mai, *ACS Appl. Mater. Interfaces*, 2015, **7**, 20902–20908.
- 36 W. Li, J. Huang, L. Cao, L. Feng and C. Yao, *Electrochim. Acta*, 2018, **274**, 334–342.
- 37 W. Li, J. Huang, L. Feng, L. Cao and S. He, *Nanoscale*, 2018, **10**, 21671–21680.
- 38 X. Zhang, Q. He, X. Xu, T. Xiong, Z. Xiao, J. Meng, X. Wang, L. Wu, J. Chen and L. Mai, *Adv. Energy Mater.*, 2020, **10**, 1904118.
- 39 Y. Yi, X. Du, Z. Zhao, Y. Liu, H. Guan, X. Liu, X. Pei, S. Zhang and D. Li, *ACS Nano*, 2022, **16**, 7772–7782.
- 40 K. H. Kim, J. Choi and S. H. Hong, *Chem. Commun. (Camb)*, 2019, **55**, 3207–3210.
- 41 Y. Zeng, J. Yang, H. Yang, Y. Yang and J. Zhao, *ACS Energy Lett.*, 2024, **9**, 1184–1191.
- 42 J. Song, B. Xiao, Y. Lin, K. Xu and X. Li, *Adv. Energy Mater.*, 2018, **8**, 1703082.

