



Cite this: *Nanoscale*, 2022, **14**, 2475

Norbornane-based covalent organic frameworks for gas separation†

Sushil Kumar, ^{‡a} Mahmoud A. Abdulhamid, ^{‡a} A. D. Dinga Wonanke,^b Matthew A. Addicoat ^b and Gyorgy Szekely ^{★a}

Covalent organic frameworks (COFs) have emerged as a new class of crystalline porous materials with distinct structural features, such as uniform pore distribution, tunable architecture, and modifiable skeletons. COFs hold significant promise for application in gas separation because of their high Brunauer–Emmett–Teller surface area and narrow pore-size distribution, which enable selective separation. The porosity and separation performance of COFs have been finely tuned by structurally modifying the starting materials. Along this direction, for the first time, we prepared W-shaped diamines by catalytic arene–norbornene annulation (CANAL) and then treated them with trialdehyde (**Tp**) to synthesize novel β -ketoenamine-linked norbornane-based COFs, *i.e.*, **ND-COF-1** and **ND-COF-2**, via a solvothermal Schiff-base condensation approach. The pore interior was decorated with methyl groups attached to the norbornane unit of the COF skeleton. Both COFs exhibited high chemical stability in different organic solvents and acidic media. Additionally, they showed high CO₂/N₂ selectivity compared with those of previously reported COFs. Moreover, their CH₄/N₂ separation efficiency was investigated, and the results revealed that **ND-COF-1** is more selective than **ND-COF-2**, which could be attributed to the less hindered pathway offered to methane gas molecules by the framework pore.

Received 17th November 2021,
Accepted 19th January 2022

DOI: 10.1039/d1nr07593d

rsc.li/nanoscale

Introduction

In the past decade, various porous materials, such as metal-organic frameworks, covalent organic frameworks (COFs), zeolites, and porous polymers, have emerged with significant application potential in a wide range of areas, including gas separation, CO₂ capture and storage, sensors, catalysis, and solar cells.¹ The need for new porous materials with unique and tailored properties is becoming increasingly important. Porous materials have been used in applications such as CO₂/N₂ gas separation from flue gas and CO₂/CH₄ separation from natural gas. Generally, to achieve high gas-pair selectivity, the chemical structure of the materials are altered through functionalization, crosslinking, thermal annealing, and carbonization to improve their size-sieving properties.² Porous polymers have amorphous morphologies, which promote non-uniform pore distributions with pore-size distributions

ranging from a few Angstroms to hundreds of nanometers. In the absence of uniform connectivity among the pores, the polymers ultimately compromise the selective gas transport through their structure. Therefore, researchers have devoted significant efforts to study the structure–property relationship of porous materials and their performance. Structural functionalization can be implemented to control the porosity and boost the separation performance of polymers with intrinsic microporosity. Additionally, altering the type of kinked unit induces notable changes to the overall porosity and gas-separation performance. Thus, developing new kinked structures for porous materials has been the object of research focus recently.

The catalytic arene–norbornene annulation (CANAL) reaction was first reported by Xia and co-workers, who used readily available monomers to develop CANAL ladder polymers with high Brunauer–Emmett–Teller (BET) surface areas and tunable microporosities.³ However, the ladder polymers were mechanically fragile and, thus, not suitable for forming free-standing membranes. To overcome the fragility limitation, CANAL polyimides and Tröger's base-based polymers were developed, and these polymers exhibited excellent gas-separation performance.^{4,5} Nonetheless, all porous polymers exhibit amorphous–semicrystalline morphologies, which prevent the control of the pore size and formation of ordered structures. Therefore, the foremost objective of researchers is to enhance

^aAdvanced Membranes and Porous Materials Center, Physical Science and Engineering Division (PSE), King Abdullah University of Science and Technology (KAUST), Thuwal 23955-6900, Saudi Arabia. E-mail: gyorgy.szekely@kaust.edu.sa; Tel: +966128082769; <http://www.szekelygroup.com>

^bSchool of Science and Technology, Nottingham Trent University, Nottingham, UK

†Electronic supplementary information (ESI) available. See DOI: 10.1039/d1nr07593d

★Both authors have contributed equally.



the crystallinity and uniform distribution of pores in the structure of polymers, to increase their gas-pair selectivity, sorption efficiency, and permeability.

Unlike polymers, COFs are well known for their crystalline and porous structures, which are generated through a covalent-bond-linked assembly of organic linkers arranged in an ordered form.^{6–8} Owing to their ordered pore structure, uniform porosity, high crystallinity, high accessible surface area, and tunable pore chemistry, COFs have diverse applications, including in optoelectronics, separation, energy storage, sensing, catalysis, and biomedicine.⁹ Conventionally, COFs are synthesized from commercially available symmetric planar organic precursor materials. These materials allow sufficient stacking between the adjacent COF layers *via* π - π interactions, which often results in the formation of well-defined porous and crystalline structures. Therefore, fusing kinked norbornane-based CANAL units with a COF backbone could enhance the separation performance of the resulting porous materials. However, to the best of our knowledge, the development of two-dimensional (2D) COFs using kinked norbornane-based CANAL ligands for gas-separation applications has not yet been reported.^{6,10,11}

In this study, two new COFs, **ND-COF-1** and **ND-COF-2**, were prepared by incorporating the kinked norbornane-based CANAL diamine unit into the structures of COFs *via* a Schiff-base condensation reaction utilizing a solvothermal approach. Furthermore, we investigated the effect of the kinked CANAL unit on the crystallinity and porosity of the resulting COFs. Additionally, the performance and selectivity of the obtained COFs in gas separation were evaluated. Both ND-COFs showed a high affinity toward CO₂, and **ND-COF-1** showed higher CH₄/N₂ selectivity than **ND-COF-2**.

Experimental

Materials

The trialdehyde precursor, 1,3,5-triformylphloroglucinol (**Tp**), was synthesized by following a previously reported procedure.¹² The materials, 4-bromo-3-methylaniline (97%), 4-bromo-2,5-dimethylaniline (99%), palladium(II) acetate (99.9%), and anhydrous 1,4-dioxane (99.8%), were purchased from Merck. Caesium carbonate (99%) was purchased from Alfa Aesar. All reagents were used as received without further purification.

Ligand syntheses

Two kinked norbornane-based CANAL diamines were prepared by following a previously reported procedure,⁴ using the CANAL reaction. The commercially obtained monomers, *i.e.*, 4-bromo-3-methylaniline and 4-bromo-2,5-dimethylaniline, were reacted with palladium(II) acetate [Pd(OAc)₂, 2 mol%], triphenylphosphine (PPh₃, 4 mol%), and caesium carbonate (Cs₂CO₃, 1 equiv.) in dry 1,4-dioxane heated at 150 °C for 24 h to afford pure products in 70%–75% yields. The two bromoaniline monomers with the *ortho* methyl group were selected to

achieve high yields and selectivity for the CANAL ligands. The chemical structures of the obtained diamines were confirmed by proton nuclear magnetic resonance (¹H NMR), ¹³C NMR, Fourier-transform infrared (FTIR) spectroscopy, and high-resolution mass spectrometry measurements.

COF syntheses

ND-COF-1 and **ND-COF-2** were synthesized in a Pyrex tube *via* the solvothermal Schiff-base condensation reaction of **Tp** (63 mg, 0.3 mmol) with the corresponding diamines in the presence of 3 M acetic acid solution (0.3 mL), using 4 mL of a 1:1 (v/v) mixture of mesitylene and dioxane as the reaction medium (Fig. 1). Subsequently, three consecutive freeze-pump-thaw cycles were performed. After sealing under vacuum, the tubes were heated to 120 °C for 72 h. **ND-COF-1** and **ND-COF-2** were precipitated as yellow solids, after which they were washed with hot water and hot *N,N*-dimethylacetamide (DMAc) to remove the excess unreacted precursors and oligomers. After the final washing with acetone at room temperature, the obtained solids were dried at 90 °C overnight to afford both COFs as yellow powders.

Characterization

Powder X-ray diffraction spectra were obtained using a Bruker D8 ADVANCE instrument in the 2θ range of 2.5°–40° and used to evaluate COF crystallinity. The experiment was carried out using a Quartz holder at a scan rate of 1° min^{−1} and a step interval of 0.01°. The stretching bands of the functional groups were obtained from FTIR spectra using a Thermo Scientific Nicolet iS10 spectrometer. Solid-state ¹³C NMR spectroscopy was performed using the Bruker Avance III 400 MHz equipment. Thermal decomposition of COFs was determined by thermogravimetric analyses (TGA, 209 F1 analyzer Netzsch) under an N₂ atmosphere at a ramp rate of 10 °C min^{−1}, in a range of 30 °C–900 °C. X-ray photoelectron spectroscopy (XPS) imaging was performed on a Kratos Axis Ultra DLD electron spectrometer. A Magellan FEI 400 was used for scanning electron microscopy (SEM) measurements. The samples were treated with iridium prior to analysis to avoid surface charging. Nitrogen adsorption analyses were conducted on a Micromeritics ASAP 2420 at −196 °C. The BET surface areas of **ND-COF-1** and **ND-COF-2** were calculated. A non-localized density functional theory (NLDFT) model was utilized for the calculations of pore size distribution. CO₂, N₂, and CH₄ uptake was measured using a Micromeritics ASAP 2020 at 0 and 25 °C, under atmospheric pressure.

Results and discussion

The two kinked norbornane-based CANAL diamines, *i.e.*, CANAL(Me)₂ and CANAL(Me)₄, were prepared *via* a CANAL reaction.⁴ The resulting diamines exhibited a “W”-shaped structure (*syn*- and *anti*-isomers) with *exo* conformation exclusively.⁴ Two new porous crystalline β -ketoenamine-linked kinked norbornane COFs, *i.e.*, **ND-COF-1** and **ND-COF-2**, were



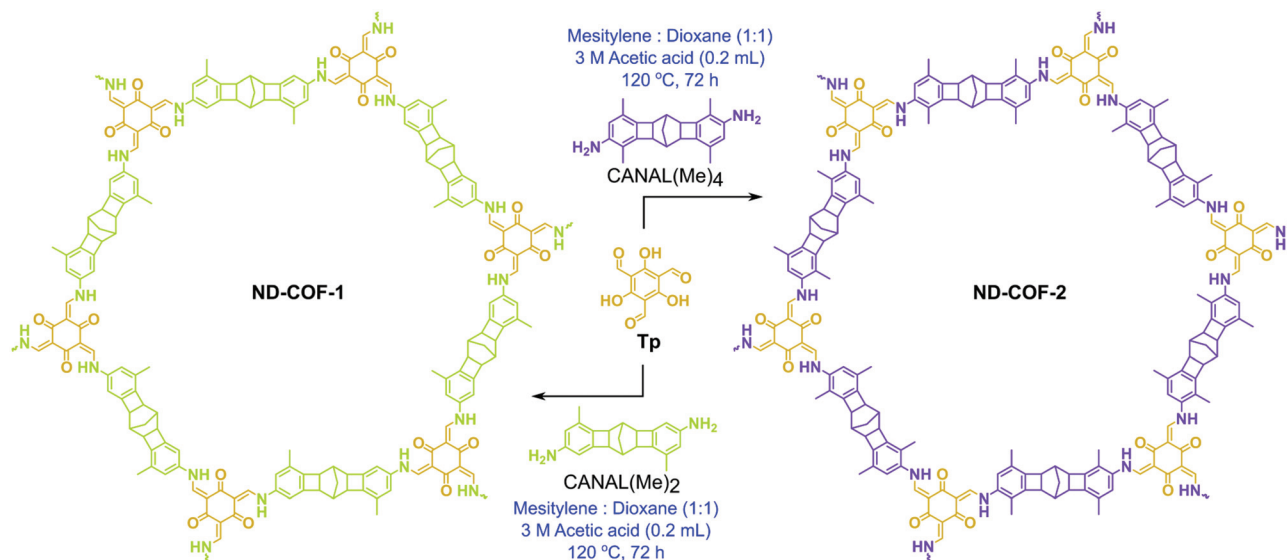


Fig. 1 Solvothermal synthesis of **ND-COF-1** and **ND-COF-2** via a Schiff-base condensation reaction involving trialdehyde (**TP**) and diamines, *i.e.*, CANAL(Me)₂ and CANAL(Me)₄, respectively.

synthesized by treating the respective kinked CANAL diamines with **TP** using a solvothermal method (Fig. 1). The ND in **ND-COF-1** and **ND-COF-2** refers to the norbornane-based diamine unit. The PXRD patterns of **ND-COF-1** and **ND-COF-2** exhibited the first high-intensity peak at $2\theta = 2.5^\circ - 10^\circ$, which corresponds to the diffraction from the [120] plane, whereas a major broad peak appeared at $2\theta = 16^\circ - 25^\circ$, assigned to the diffraction from the [322] plane. Given the three-dimensional nature of the norbornane unit and attached methyl groups, a large distance between the COF layers is expected. In fact, the sensitivity of PXRD in that region is low. During simulation, assigning a *c* lattice parameter of 12 Å, and taking into account the ABC stacking, gives an inter-linker distance of just below 5 Å (Fig. 2a and b). The relatively high intensity of the first peak indicated the crystalline nature of the ND-COFs. The experimental PXRD patterns matched well with the simulated patterns generated using the ABC stacking model.

This excellent agreement between the experimental and simulated PXRD patterns was further verified by Pawley refinement (**ND-COF-1**: $R_p = 5.96\%$, $R_{wp} = 7.23\%$ and **ND-COF-2**: $R_p = 6.74\%$, $R_{wp} = 7.76\%$), performed using the Material Studio software refinement package. The FTIR spectra confirmed the absence of any trace of the starting material in **ND-COF-1** and **ND-COF-2** (Fig. 2c, section S4 of the ESI†). This was further supported by the disappearance of the NH stretching band ($3178-3200\text{ cm}^{-1}$) of CANAL(Me)₂ and CANAL(Me)₄ diamines, as well as the C=O stretching band (1639 cm^{-1}) of **TP**. However, the carbonyl (C=O) peak broadened and merged with the C=C stretching band, as shown in Fig. 2c, which is indicative of the extended framework structure. The appearance of the C=C stretching band at approx. 1578 cm^{-1} and a strong band at approx. 1260 cm^{-1} affirms the formation of β -ketoenamine linkages between **TP** and the respective diamine precursors in the COFs.

The chemical structures of the COFs were determined from the XPS survey profiles (section S5, ESI†). The XPS full survey profile of both COFs shows three intense peaks, assigned to C (1s), O (1s), and N (1s) signals, respectively. For **ND-COF-1**, the deconvoluted XPS profile for C 1s showed a signal at 284.27 eV, assigned to the C=C moiety of the aromatic ring system. Moreover, the shoulders appearing at 285.13 and 287.21 eV were assigned to the C-O and C=O bonds, respectively. In addition, a satellite peak was observed at 290.74 eV, assigned to the $\pi-\pi^*$ transition. The high-resolution spectrum of N (1s) showed one intense signal at 399.52 eV, corresponding to the enamine nitrogen of the C=C-NH- bond formed *via* tautomerism during the condensation reaction. The deconvoluted XPS profile of O 1s showed a peak at 530.34 eV, assigned to the C=O group. The shoulder peak appearing at 532.19 eV was assigned to the C-O bonds of the framework. For **ND-COF-2**, the deconvoluted XPS profile of C 1s showed a signal at 284.24 eV, assigned to the C=C moiety of the aromatic ring system. The shoulder peaks at 285.48 and 286.35 eV were assigned to the C-O and C=O bonds of the framework, respectively. In addition, the satellite peak at 290.37 eV was attributed to the $\pi-\pi^*$ transition. The N 1s XPS profile showed a signal at 399.53 eV, assigned to the C=C-NH- bond in the framework. The additional signal at 400.87 eV is most probably derived from contamination by another diamine isomer present in small amounts in the COF structure (section 5, ESI†). The deconvoluted XPS profile of O 1s showed a peak at 530.39 eV (for the C=O bond) and a shoulder at 532.0 eV, assigned to the C-O bonds in the framework.

The formation of the β -ketoenamine linkage in the framework structure was further verified by solid-state NMR spectroscopy. The ^{13}C solid-state CP-MAS NMR spectra exhibited intense peaks at 184.59 and 184.54 ppm for **ND-COF-1** and **ND-COF-2**, respectively. These peaks can be assigned to the



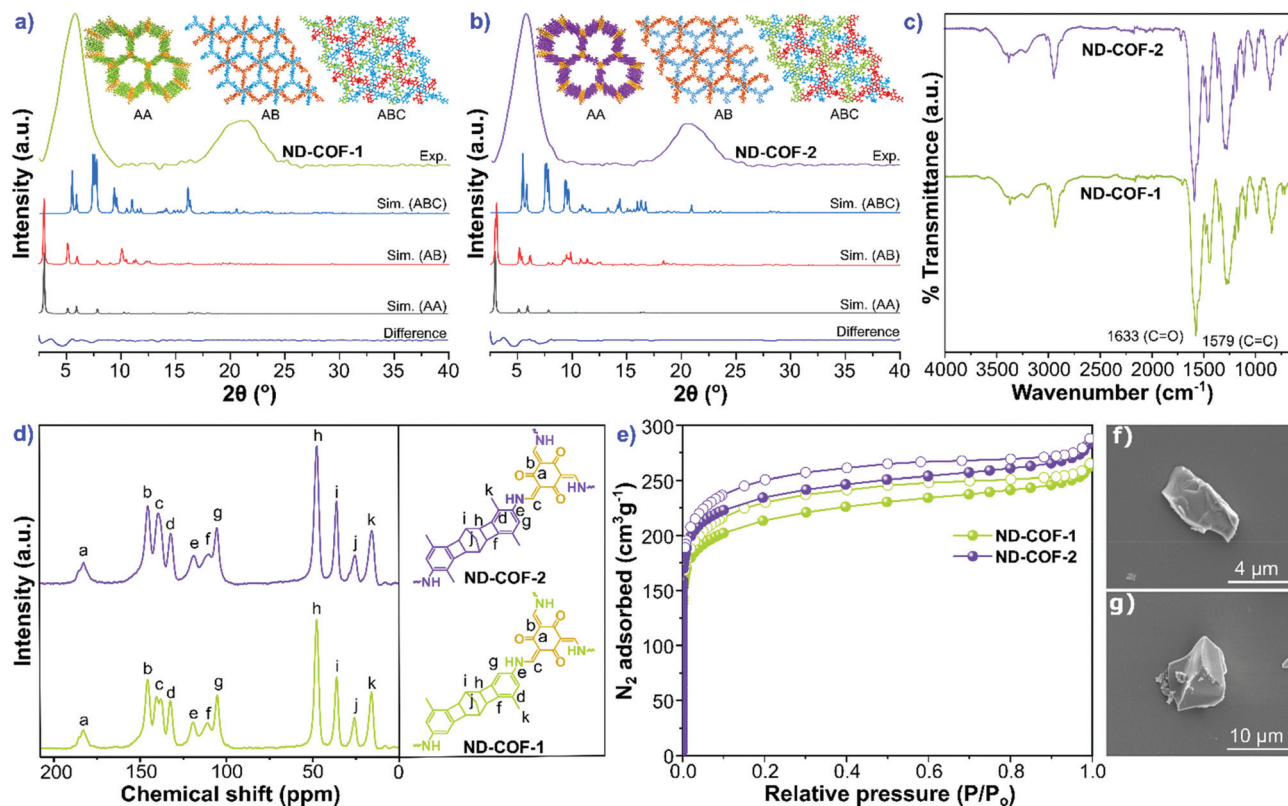


Fig. 2 Experimental and simulated PXRD patterns of (a) ND-COF-1 and (b) ND-COF-2, (c) FTIR spectra of both ND-COFs, (d) solid-state ^{13}C NMR spectra of ND-COF-1 and ND-COF-2, (e) N_2 adsorption and desorption isotherms of the ND-COFs recorded at 77 K, and SEM images of (f) ND-COF-1 and (g) ND-COF-2.

carbonyl ($\text{C}=\text{O}$) group of the β -ketoenamine linkage (Fig. 2d). Conventionally, the carbonyl group ($\text{C}=\text{O}$) of **Tp** resonates downfield at 192 ppm.¹³ An upfield shift of the $\text{C}=\text{O}$ peak to 184 ppm for the ND-COFs suggests the formation of a keto group in the framework. In addition, the absence of the characteristic $\text{C}=\text{O}$ peak (assigned to the aldehyde moiety present in **Tp**) at 192 ppm in the NMR spectra of the ND-COFs suggests the total consumption of the **Tp** precursor. The NMR peaks resonating in the range of 100–150 ppm can be assigned to the carbon atoms of the aromatic ring. The NMR peaks appearing in the range of 22–55 ppm were assigned to the carbon signals of the norbornane moiety in the framework. The peak resonating at approx. 16 ppm corresponds to the sp^3 carbon (methyl group) of the COFs.

The permanent porosities of ND-COF-1 and ND-COF-2 were determined by measuring the nitrogen gas uptake isotherm at -196°C (Fig. 2e). Both COFs exhibited the type-I gas adsorption isotherm, which indicates their microporous nature. The BET surface areas of activated ND-COF-1 and ND-COF-2 were determined to be 794 and 879 $\text{m}^2 \text{g}^{-1}$, respectively, which are similar values. However, the calculated surface area using a 1.86 Å radius probe to model N_2 is quite sensitive to the position of the methyl groups, because the methyl groups may create triangular pockets in ABC stacking. Most probably, ND-COF-2 has somewhat better crystallinity, which resulted in

the increase in the surface area compared to ND-COF-1. A comparison of theoretical and experimental surface areas calculated for both COFs is given in Table S7, ESI†. Due to the extended length of the organic diamine linkers, the iso-reticulated structures of the ND-COFs exhibited pore apertures lower than 20 Å. Using the NLDFT method, the pore-size distributions of both ND-COF-1 and ND-COF-2 were calculated to be approx. 16.3 Å and 14.5 Å, respectively. The pore size values calculated from the ABC crystal model were found to be close to the experimental values (Fig. S19 and S20, ESI†). The SEM images revealed that ND-COF-1 and ND-COF-2 crystallized in a sheet-like morphology (Fig. 2f and g). TGA of activated ND-COF-1 and ND-COF-2 was performed to determine the thermal stability of the framework and the presence of any guest molecule in the pores (section S6, ESI†). Both COFs showed high thermal stability up to approx. 380 °C. A gradual weight loss after 380 °C was observed, attributable to the decomposition of the COF structure.

To investigate the chemical stabilities of ND-COF-1 and ND-COF-2 in solvents, they were immersed in different organic solvents, including DMAc, methanol, tetrahydrofuran (THF), hexane, and water, for seven days (section S8, ESI†). Both COFs were stable in these solvents. The crystallinity of the treated COFs was examined by PXRD (section S8, ESI†). The absence of significant changes in the relative peak intensities and peak



positions in the PXRD patterns of the respective COFs indicates their high chemical stability in the examined solvents, including water. A similar observation was made in the FTIR spectra of the treated COFs, where no significant changes in the characteristic FT-IR stretches were observed (section S8, ESI†).

The stability of **ND-COF-1** and **ND-COF-2** was further examined in acids (section S8, ESI†). Both COFs were submerged in 5 mL of 3 M HCl solution for seven days. The PXRD patterns of both acid-treated COFs showed no noticeable variation in the relative peak intensities and positions, which indicated the high chemical stability of both COFs. Similarly, no significant variation in the IR stretches was observed, indicating the intactness of the framework even in acidic environments. Furthermore, we evaluated the chemical stability of both COFs under alkaline conditions (section S8, ESI†). Both COFs were treated with 3 M NaOH solution for seven days. The PXRD pattern of **ND-COF-1** showed no significant variation when compared to that of the pristine COF. Conversely, the PXRD pattern of **ND-COF-2** exhibited an additional signal along with the characteristic COF peak, which indicated the disintegration of the framework. Thus, **ND-COF-1** is more stable in alkaline media than **ND-COF-2**. The FTIR spectra of the treated COFs further support this result.

To explore the separation performance of the COFs, we conducted sorption experiments at 1 bar with three different gases: CO₂, N₂, and CH₄. The adsorption isotherms of CO₂, N₂, and CH₄ measured at 0 °C and 25 °C are presented in Fig. 3.

Both **ND-COF-1** and **ND-COF-2** demonstrated high CO₂ adsorption capacities of 2.24 and 2.12 mmol g⁻¹, respectively, at 0 °C and 1 bar (Fig. 3). Interestingly, **ND-COF-1** exhibited similar CO₂ and CH₄ sorption capacities (Fig. 3a–c), whereas the CO₂ sorption capacity of **ND-COF-2** was 100% higher than that for CH₄ (Fig. 3d–f). The high CO₂ adsorption performance of **ND-COF-2** relative to that of **ND-COF-1** is attributable to the higher surface area and somewhat better crystallinity of **ND-COF-2**. Notably, the N₂ sorption capacity at room temperature under atmospheric pressure is marginal because of its low critical temperature (low condensability).

The effect of temperature on the COF separation performance was evaluated at 0 °C and 25 °C at 1 bar. In particular, a 59% reduction in the CO₂ sorption capacity of **ND-COF-1** was observed when the temperature was increased from 0 °C to 25 °C (Fig. 3a and b). This corresponded to the change from 2.245 to 1.416 mmol g⁻¹. However, for **ND-COF-2**, a 48% reduction in the CO₂ sorption capacity was observed, corresponding to the change from 2.124 to 1.428 mmol g⁻¹ (Fig. 3d and e). Under the same conditions, both COFs exhibited high percentage reductions in N₂ and CH₄ sorption capacities compared to their CO₂ sorption capacities. In particular, **ND-COF-1** exhibited 59% and 83% reductions in N₂ and CH₄ sorption capacities, respectively, whereas **ND-COF-2** showed 275% and 54% reductions, respectively. Table 1 shows the CO₂ sorption capacities of **ND-COF-1** and **ND-COF-2** compared with those of the **Tp**-based COFs reported in the literature. The obtained CO₂ sorption capacities of **ND-COF-1** and **ND-COF-2** were

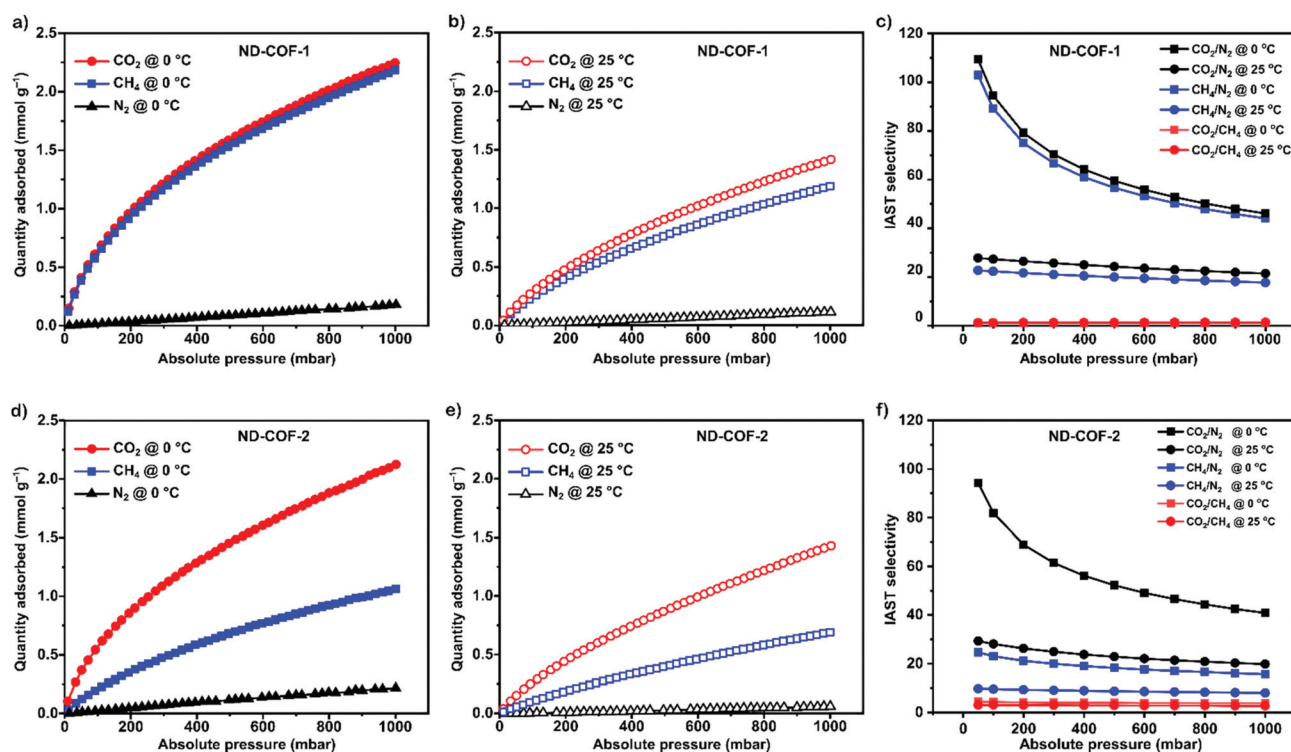
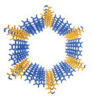
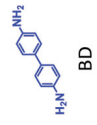
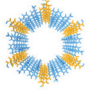
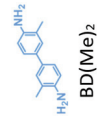
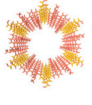
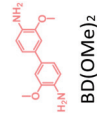
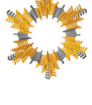
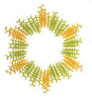
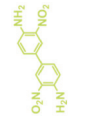
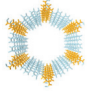
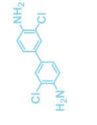
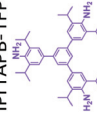
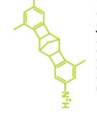
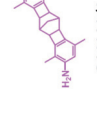


Fig. 3 Gas adsorption and selectivity data for the ND-COFs. **ND-COF-1**: CO₂, CH₄, and N₂ adsorption isotherms at (a) 0 °C and (b) 25 °C; (c) CO₂/N₂, CH₄/N₂, and CO₂/CH₄ ideal adsorbed solution theory (IAST) selectivity at 0 °C and 25 °C. **ND-COF-2**: CO₂, CH₄, and N₂ adsorption isotherm at (d) 0 °C and (e) 25 °C; (f) CO₂/N₂, CH₄/N₂, and CO₂/CH₄ IAST selectivity at 0 °C and 25 °C.



Table 1 CO₂ sorption capacities of ND-COF-1 and ND-COF-2 compared with those of different Tp-based COFs reported in the literature

COF structure	Amine unit	SA _{BET} (m ² g ⁻¹)	Pore width (Å)	CO ₂ sorption (mmol g ⁻¹)	Ref.
		35	17.2	1.92	14
		468	14.2	1.65	13
		330	14.1	1.2	13
		415	9.6	0.97	15
		295	16.8	2.32	13
		723	18.6	2.67	16
		756	34	2.6	17
		794	30.9	2.24	This work
		879	30.9	2.12	This work

higher than those of the COFs obtained using diamines such as benzidine (**TpBD**; 1.92 mmol g⁻¹),¹⁴ *ortho*-toluidine (**TpBD(Me)**; 1.65 mmol g⁻¹),¹³ *ortho*-toluidine (**TpBD(OMe)**; 1.2 mmol g⁻¹),¹³ and hydrazine units (**COF-JLU2**; 0.97 mmol g⁻¹).¹⁵ However, the CO₂ sorption capacities of **ND-COF-1** and **ND-COF-2** were comparable to those of other COFs prepared using diamines such as dinitrobenzidine (**TpBD(NO₂)₂**; 2.32 mmol g⁻¹),¹³ BD(Cl)₂ (**CAA-COF-2**; 2.67 mmol g⁻¹),¹⁶ and triamine (e.g., TAPB) (**iPrTAPB-TFP**, 2.6 g⁻¹).¹⁷ The gas-pair selectivities of both COFs were calculated using the ideal adsorbed solution theory (IAST) at a partial equilibrium pressure of 0.5 bar/0.5 bar (Fig. 3c and f). **ND-COF-1** and **ND-COF-2** exhibited high CO₂/N₂ selectivities of 46 and 41 at 0 °C and 1 bar, respectively. These selectivities were reduced by approximately 50% upon increasing the temperature to 25 °C under atmospheric pressure. Remarkably, **ND-COF-1** showed an excellent CH₄/N₂ selectivity of 43 at 0 °C, which is 175% higher than that exhibited by **ND-COF-2**. The performance results show the potential of employing kinked norbornane-based COFs in industrial applications, exploiting their ease of preparation, high sorption capacity, and high gas-pair selectivity.

Conclusions

In summary, we have synthesized crystalline and porous β-ketoenamine-linked kinked norbornane COFs (**ND-COF-1** and **ND-COF-2**) and investigated the effect of kinked CANAL-unit-decorated frameworks in gas separation. Both COFs exhibited high chemical stability in different organic polar and non-polar solvents, including acidic media. However, **ND-COF-2** disintegrated in alkaline media, whereas **ND-COF-1** remained stable. Interestingly, due to the presence of a methyl-functional-group-decorated pore interior, the CO₂ adsorption efficiency of **ND-COF-2** was approximately 100% higher than its CH₄ sorption efficiency. According to the IAST calculations, **ND-COF-1** showed 175% higher CH₄/N₂ selectivity than **ND-COF-2** at 0 °C and 1 bar. In future studies, new designs of CANAL-based linkers through pre-synthetic or post-synthetic functionalization will be explored to improve the porosity and selectivity of the obtained COFs, thereby enhancing their separation performance.

Author contributions

SK and MAH have contributed equally. MAH: Ligand design, investigation, validation, analysis, data curation, gas separation, and writing – original draft. SK: COFs – design, synthesis, characterization, investigation, validation, formal analysis, data curation, visualization, refinement, and writing – original draft. AD: Modeling, simulation, and computational analysis. MAA: Modeling, simulation, computational analysis, and supervision. GS: Conceptualization, methodology, resources, visualization, writing – review & editing, supervision, funding acquisition, and project administration.



Conflicts of interest

There are no conflicts to declare.

Acknowledgements

SK and MAH thank King Abdullah University of Science and Technology (KAUST) for the postdoctoral fellowship. The research reported in this publication was supported by funding from KAUST. We acknowledge Rifan Hardian for the IAST calculations, Abdul Hamid Emwas for the solid-state ^{13}C CP-MAS NMR analysis, and Mohamed Nejib Hedhili for the XPS measurements. MAA thanks the Materials Chemistry Consortium for HPC access (EP/T022213).

Notes and references

- 1 M. Dincă and J. R. Long, *Chem. Rev.*, 2020, **120**, 8037–8038.
- 2 Y. Wang, B. S. Ghanem, Z. Ali, K. Hazazi, Y. Han and I. Pinnau, *Small Struct.*, 2021, **2**, 2100049.
- 3 H. W. H. Lai, S. Liu and Y. Xia, *J. Polym. Sci., Part A: Polym. Chem.*, 2017, **55**, 3075–3081.
- 4 M. A. Abdulhamid, H. W. H. Lai, Y. Wang, Z. Jin, Y. C. Teo, X. Ma, I. Pinnau and Y. Xia, *Chem. Mater.*, 2019, **31**, 1767–1774.
- 5 X. Ma, H. W. H. Lai, Y. Wang, A. Alhazmi, Y. Xia and I. Pinnau, *ACS Macro Lett.*, 2020, **9**, 680–685.
- 6 S. J. Lyle, P. J. Waller and O. M. Yaghi, *Trends Chem.*, 2019, **1**, 172–184.
- 7 K. Dey, S. Mohata and R. Banerjee, *ACS Nano*, 2021, **15**, 12723–12740.
- 8 M. S. Lohse and T. Bein, *Adv. Funct. Mater.*, 2018, **28**, 1705553.
- 9 K. Geng, T. He, R. Liu, S. Dalapati, K. T. Tan, Z. Li, S. Tao, Y. Gong, Q. Jiang and D. Jiang, *Chem. Rev.*, 2020, **120**, 8814–8933.
- 10 Y. Wang, C. Wu, W. Sun, Q. Pan, W. Hao, H. Liu, J. Sun, Z. Li, J. Sun and Y. Zhao, *Mater. Chem. Front.*, 2021, **5**, 944–949.
- 11 H. Li, F. Chen, X. Guan, J. Li, C. Li, B. Tang, V. Valtchev, Y. Yan, S. Qiu and Q. Fang, *J. Am. Chem. Soc.*, 2021, **143**, 2654–2659.
- 12 J. H. Chong, M. Sauer, B. O. Patrick and M. J. MacLachlan, *Org. Lett.*, 2003, **5**, 3823–3826.
- 13 S. Chandra, S. Kandambeth, B. P. Biswal, B. Lukose, S. M. Kunjir, M. Chaudhary, R. Babarao, T. Heine and R. Banerjee, *J. Am. Chem. Soc.*, 2013, **135**, 17853–17861.
- 14 B. P. Biswal, S. Chandra, S. Kandambeth, B. Lukose, T. Heine and R. Banerjee, *J. Am. Chem. Soc.*, 2013, **135**, 5328–5331.
- 15 Z. Li, Y. Zhi, X. Feng, X. Ding, Y. Zou, X. Liu and Y. Mu, *Chem. – Eur. J.*, 2015, **21**, 12079–12084.
- 16 D. B. Shinde, M. Ostwal, X. Wang, A. M. Hengne, Y. Liu, G. Sheng, K.-W. Huang and Z. Lai, *CrystEngComm*, 2018, **20**, 7621–7625.
- 17 D. Kaleeswaran, P. Vishnoi and R. Murugavel, *J. Mater. Chem. C*, 2015, **3**, 7159–7171.

