

## Surfactant-assisted synthesis of $\text{CeO}_2$ nanoparticles and their application in wastewater treatment<sup>†</sup>

He Li,<sup>a</sup> Guofeng Wang,<sup>a</sup> Fei Zhang,<sup>a</sup> Yun Cai,<sup>a</sup> Yude Wang<sup>\*a</sup> and Igor Djerdj<sup>\*b</sup>

Received 6th June 2012, Accepted 12th October 2012

DOI: 10.1039/c2ra21590j

The stable and crystalline phase of pure nanostructured  $\text{CeO}_2$  with various morphologies has been directly synthesized using a cationic surfactant (cetyltrimethylammonium bromide, CTAB) and cerium chloride ( $\text{CeCl}_3 \cdot 6\text{H}_2\text{O}$ ) at room temperature by a new, simple, and green chemical precipitation method. Thorough structural characterization techniques, including X-ray diffraction (XRD), Fourier transform infrared spectroscopy (FTIR), scanning electron microscopy (SEM), transmission electron microscopy (TEM), electron diffraction, and Raman spectroscopy, were employed to examine the morphology and the microstructure of the final product. The catalytic activity of the nanostructured  $\text{CeO}_2$  was tested towards the degradation of the azo dye Congo red (CR). In order to obtain the optimum degradation conditions of CR, the performance of nanostructured  $\text{CeO}_2$  with various morphologies (spherical nanoparticles, nanorods, and mixture thereof) for the removal of CR from wastewater was tested under various concentrations of CR dye and quantities of the nanostructured  $\text{CeO}_2$ . The results show an excellent removal capacity for the organic pollutant CR from wastewater, making it a promising candidate for wastewater treatment.

### 1. Introduction

As one of the most reactive rare earth oxides, with a wide bandgap ( $E_g = 3.15$  eV for the coarse-grained material) and high dielectric constant ( $\epsilon = 24.5$ ),<sup>1</sup> ceria (cerium oxide,  $\text{CeO}_2$ ) has attracted intense interest because it plays a vital role in emerging technologies for environmental and energy related applications, such as heterogeneous catalysis,<sup>2</sup> solid oxide fuel cells (SOFCs),<sup>3</sup> optics,<sup>4</sup> polishing materials,<sup>5</sup> gas sensors,<sup>6</sup> UV light blockers,<sup>7</sup> and especially in three-way catalysts (TWCs).<sup>8–10</sup> Moreover, nanostructured  $\text{CeO}_2$  materials have received tremendous attention because of their unique properties which are derived from their size, shape, orientation, and high surface area. Nanostructured ceria with various morphologies (spherical nanoparticles and one-dimensional nanostructures) exhibits excellent physicochemical properties, resulting in its important applications as three-way catalysts in vehicle emission control systems, electrolyte materials for solid oxide fuel cells, ultraviolet-blocking materials, conversion catalysts, solar cells, gates for metal-oxide semiconductor devices, polishing materials, sunscreen cosmetics, etc.<sup>6,11–16</sup> Motivated by both the possible applications and the excellent properties, the development of efficient methods to synthesize nanostructures with size/shape-

control is one of the key trends in materials science nowadays. Over the past few years, the syntheses of size/shape-controlled  $\text{CeO}_2$  nanostructures, such as spherical nanoparticles, nanorods, nanowires, nanotubes, nanocubes, nanospheres, and hierarchical nanostructures, accompanied by the investigation of their size/shape-dependent properties, have been carried out by ourselves and other research groups.<sup>17–27</sup> However, these methods have defects in the complex process and expensive raw materials, or require very stringent control of various processing parameters, together with a low production yield, high energy consumption, complicated manufacture, and a rather not environmentally friendly behaviour, the cost of the production is very high and it is difficult to be realized at an industrial scale. In order to overcome these problems, the simple, environmentally benign, and low cost routes to synthesize nanostructured  $\text{CeO}_2$  with different morphologies are still the key issues and challenge, especially in the viewpoints of “green chemistry”. Kluson and co-workers have successfully prepared titania nanosized particles by a surfactant-mediated sol–gel method from a non-ionic surfactant and titanium(IV) isopropoxide.<sup>28</sup> The synthesis of cerium oxides has been achieved *via* surfactant-templated synthesis.<sup>29</sup> In these methods, it is necessary for samples to withstand post-calcination at high temperature, so that the crystalline  $\text{CeO}_2$  can be obtained. Our method of the controllable synthesis of  $\text{CeO}_2$  nanostructures is based on the cationic surfactant CTAB and simple chemical reagents hydrous cerium chloride ( $\text{CeCl}_3 \cdot 6\text{H}_2\text{O}$  and  $\text{NH}_3 \cdot \text{H}_2\text{O}$ ) at room temperature by controlling the molar ratio of CTAB to cerium species and the aging times in an aqueous solution. It is a low energy consuming

<sup>a</sup>Department of Materials Science & Engineering, Yunnan University, 650091 Kunming, People's Republic of China.

E-mail: ydwang@ynu.edu.cn; Fax: +868715031410; Tel: +868715031124

<sup>b</sup>Ruder Bošković Institute, Bijenička 54, 10000 Zagreb, Croatia.

E-mail: igor.djerdj@irb.hr; Fax: +38514680114; Tel: +38514680113

† Electronic Supplementary Information (ESI) available: Fig. S1–S9. See DOI: 10.1039/c2ra21590j

and environmentally benign green process because the post-calcinations or thermal treatments at high temperature are avoided. The cations (cerium) are assembled within the template of the surfactant micelle in an aqueous solution. The surfactant not only provides a favorable site for the growth of the particulate assemblies, it also influences the morphology and the formation process, including nucleation, growth, coagulation and flocculation.<sup>30</sup>

As an inexpensive and relatively harmless material, CeO<sub>2</sub> presents several characteristics that could be potentially advantageous for catalytic applications. It has been investigated under UV irradiation concerning water splitting for the generation of hydrogen gas,<sup>31,32</sup> and photodegradation of toluene in the gas phase.<sup>33</sup> Many researchers have reported that the CeO<sub>2</sub> nanostructures exhibit catalytic behaviors under photoirradiation to degrade dyes.<sup>34–37</sup> Generally, CeO<sub>2</sub> nanostructures have apparent advantages in enhancing the catalytic activities because such structures not only possess a high surface area but can also effectively transport reactant molecules to the active sites.<sup>38,39</sup> The removal of gaseous noxious compounds from automobile exhausts is an inevitable component and the main environmental application of ceria.<sup>40</sup> Recently, ceria could be potentially used as a near-UV-Vis range irradiation or visible light responsive catalyst to decompose organics in the aqueous phase.<sup>19,41–43</sup> However, upon comparison with the plentiful nanostructures synthesized with different methods, it appears that there is little work that has been carried out to investigate their catalytic behavior so far. With this aim, in the present work we study the catalytic degradation of organic dyes utilizing CeO<sub>2</sub> nanostructures prepared from an aqueous solution at room temperature.

Organic dye removal from wastewater has attracted considerable attention because of their long-term environmental toxicity and short-term public health damage.<sup>41,44</sup> Congo red [1-naphthalene sulfonic acid, 3,30-(4,40-biphenylenebis(azo))bis(4-amino-)disodium salt] with two azo bond (–N=N–) chromophores in its molecular structure is a well-known class of azo dyes that are of high toxicity and are even carcinogenic to animals and humans, and they are not readily degradable. CR dyes from the textile and dye industries are causing grave environmental problems and are banned in many countries because of health concerns. However, the traditional physical, chemical and biological means of wastewater treatment often have little degradation effect on this anionic secondary diazo dye pollutant due to their complex aromatic structures, which provide them with physicochemical, thermal and optical stability.<sup>45,46</sup> Compared with the conventional wastewater treatment means, degradation has been proved to be effective for this kind of pollution due to its strong destructive power to mineralize the pollutants into CO<sub>2</sub> and H<sub>2</sub>O. The present investigation assesses the applicability of materials for the degradation and removal of the CR from wastewater, such as TiO<sub>2</sub>,<sup>47–50</sup> zeolite,<sup>51</sup> Ni(OH)<sub>3</sub> and NiO,<sup>52</sup> Fe<sub>2</sub>O<sub>3</sub>,<sup>53</sup> CeO<sub>2</sub>,<sup>19,45</sup> carbon nanotubes,<sup>54</sup> CdS,<sup>55</sup> Zn<sub>1-x</sub>Cu<sub>x</sub>S and Zn<sub>1-x</sub>Ni<sub>x</sub>S,<sup>56</sup> ZnO,<sup>57</sup> WO<sub>3</sub>–TiO<sub>2</sub>/activated carbon,<sup>46</sup> SnO<sub>2</sub>–ZnO,<sup>58</sup> and waste materials (bottom ash and deoiled soya).<sup>59</sup> However, their practical uses have been constrained by their low catalytic activity under solar light, short-term stability against photo and chemical corrosion as well as potential toxicity. To our knowledge, the catalytic activity of CeO<sub>2</sub> nanostructures of various

morphologies, used as a potential catalytic material for the degradation and removal of the CR from wastewater has rarely been reported. In this paper, CR dye was selected as a model dye pollutant to evaluate the degradation activity of the CeO<sub>2</sub> nanostructures. The obtained data are discussed in connection with the differences in the activity of the CeO<sub>2</sub> nanostructures.

## 2. Experimental

### 2.1 Synthesis of CeO<sub>2</sub> nanostructures

All the chemical reagents used in the experiments were obtained from commercial sources as guaranteed-grade reagents and used without further purification. The purity of CTAB was 99% and of the inorganic precursors was not lower than 99%.

The synthesis was based on the use of the cationic surfactant (CTAB) and the simple chemical materials (cerium chloride CeCl<sub>3</sub> and NH<sub>3</sub>·H<sub>2</sub>O) as inorganic precursors. The reaction was performed at room temperature. The starting compositions and synthesis conditions are presented in Table 1. In a typical process of CeO<sub>2</sub> nanoparticle preparation, the synthetic procedures were as follows: the CTAB was mixed with distilled deionized water under stirring until a homogeneous solution (0.1 M) was obtained. The solution of diluted NH<sub>3</sub>·H<sub>2</sub>O (25 wt% solution, 10 mL) was then added into the CTAB solution under vigorous stirring. When the mixing solution became homogeneous, a solution of CeCl<sub>3</sub> (0.1 M) was added, also under stirring condition. After stirring for 2 h, the products were aged at ambient temperature for 30 min. The resulting material was filtered, washed with distilled water to remove surfactant, and then dried at ambient temperature.

### 2.2 Characterization of the CeO<sub>2</sub> nanostructures

Powder X-ray diffraction (XRD) data was recorded with a Rigaku D/MAX-3B powder diffractometer using the copper target and K<sub>α</sub> radiation ( $\lambda = 1.54056 \text{ \AA}$ ). The sample was scanned from 20° to 90° (2 $θ$ ) in steps of 0.02°. The Rietveld refinement of the powder XRD patterns was performed using the programme FULLPROF.<sup>60</sup> The profile function was chosen to be the modified Thompson–Cox–Hastings pseudo-Voigt (TCH pV), allowing the straightforward size analysis. In this approach, we assumed that the line-broadening of the deconvoluted profile occurred due to the small crystallite size only. The values of half-width parameters  $U$ ,  $V$ ,  $W$ , and  $X$  were kept constant at instrumental values determined by using the standard sample LaB<sub>6</sub>. Fourier transform infrared (FT-IR) spectroscopy was performed on a Perkin-Elmer 2000 FT-IR spectrometer. Scanning electron microscopy (SEM) images of the final particles were obtained with an XL30ESEM-TMP microscope. The samples for SEM were prepared by dispersing the final powders in the conductive glue; this dispersion was then sprayed with carbon. Transmission electron microscopy (TEM) measurements were performed on a Zeiss EM 912 Ω instrument at an acceleration voltage of 120 kV, while high-resolution transmission electron microscopy (HRTEM) characterization was done using a JEOL JEM-2010 Electron Microscope (with an acceleration voltage of 200 kV). The samples for TEM were prepared by dispersing the final samples in distilled deionized water; this dispersion was then dropped on the carbon–copper grids coated

**Table 1** Morphologies, crystallite sizes and BET surface areas of the CeO<sub>2</sub> nanostructures synthesized under various experimental conditions

CTAB : Ce <sup>3+</sup> (molar ratio)	Aging time	Observed morphology	Lattice parameter <sup>a</sup> (Å)	Average crystallite size <sup>b</sup> (nm)	S <sub>BET</sub> <sup>c</sup> (m <sup>2</sup> g <sup>-1</sup> )
1.0 : 1.0	30 min	Spherical nanoparticles	5.413(1)	5.0	127.8
1.5 : 1.0	10 days	Nanorods	5.414(1)	4.3	132.9
1.0 : 1.0	10 days	Spherical nanoparticles and nanorods	5.414(1)	4.4	121.1

<sup>a</sup> The lattice parameter was extracted using the Rietveld refinement. <sup>b</sup> The average crystallite size ( $D$ ) of the as-prepared CeO<sub>2</sub> nanoparticles was calculated from the peak broadening using the Rietveld method. <sup>c</sup> BET specific surface area.

by an amorphous carbon film. N<sub>2</sub> adsorption–desorption isotherms at 77 K were recorded on a Micromeritics ASAP 2010 automated sorption analyzer. The samples were outgassed at 150 °C for 20 h before sorption analysis. UV-Vis measurements were made with a UV-2401PC spectrophotometer.

### 2.3 Water treatment with CeO<sub>2</sub> nanostructures

CR (C<sub>32</sub>H<sub>22</sub>N<sub>6</sub>Na<sub>2</sub>O<sub>6</sub>S<sub>2</sub>, molecular weight: 696.67 g mol<sup>-1</sup>, from Guangzhou Reagent Corporation, China) was used as a model dye to investigate the catalytic activity of the CeO<sub>2</sub> nanostructures. The degradation was carried out in an aqueous solution at ambient temperature using CeO<sub>2</sub> nanostructures as the catalyst. The amount of CeO<sub>2</sub> nanostructures contained in the dye solution, initial CR concentration and irradiation time were varied to determine the optimum working conditions for CR degradation. In a typical degradation experiment, the proper amounts (20, 50, 100, 150, 200, and 250 mg) of the CeO<sub>2</sub> nanostructures were added to 50 mL of CR aqueous solution (CRS) with different concentrations (10, 30, 50, 80, and 100 mg L<sup>-1</sup>) in a quartz beaker under stirring. 4 mL of the dispersion was extracted and subsequently centrifuged to separate CeO<sub>2</sub> nanostructures and dye solutions at 3500 rpm for 10 min at different intervals. The changes of the concentration of the centrifuged solutions were recorded by a UV-2401PC spectrophotometer. The concentration ( $C$ ) of the centrifuged solution and the initial concentration ( $C_0$ ) of the CR solution were monitored immediately by measuring the absorbance of the supernatant at 498 nm using a spectrophotometer (model no. JH722N), and was finally analyzed by a UV-2401PC spectrophotometer. The degradation rate was calculated as  $C/C_0$ . CeO<sub>2</sub> powders were thoroughly washed, dried and then used as the recycled catalysts to degrade new CR solutions. The degradation products formed at the end of the degradation reaction process were analyzed by liquid chromatography–mass spectrometry (LC–MS) measurements performed using Agilent G1969A (Agilent Technologies Inc. USA) with electrospray ionization mass spectrometry.

## 3. Results and discussion

### 3.1 Physicochemical characterization

**3.1.1 Structural characterization.** The XRD patterns of the three types of the final powders (Fig. 1) reveal well-developed reflections of cerium oxide (ICDD PDF No. 81-0792), space group *Fm3m* (225). No crystalline by-products, such as CTAB, CeCl<sub>3</sub> or other cerium oxides, were found in the pattern, indicating that the as-synthesized samples are phase pure CeO<sub>2</sub> with cubic fluorite structure. The main structural parameters obtained from the Rietveld refinement of the powder XRD

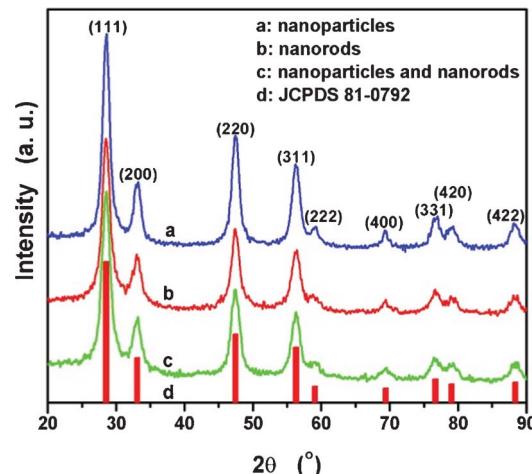


Fig. 1 XRD patterns of as-synthesized CeO<sub>2</sub> nanostructures.

patterns are presented in Table 1. The obtained average crystallite sizes ( $D$ ) of the as-prepared CeO<sub>2</sub> nanoparticles/nanorods show no anisotropical behavior (no  $hkl$  dependence) and lie in the narrow range between 4.3 and 5 nm. Unit cell parameter values for these CeO<sub>2</sub> nanostructures reveals that the unit cell experiences some expansion with respect to that of the bulk cerium oxide ( $a = 5.409$  Å). As a general rule, nanoparticles of oxides exhibit a lattice expansion, which can be assigned to lower size ceria nanoparticles and are related to the presence of Ce<sup>3+</sup> and oxygen vacancies in an amount increasing with the decreasing of the particle size.<sup>61</sup> The expansion of the lattice parameter can be related to the higher surface-to-volume ratio in the smaller particles, resulting in a higher contribution from the surface layer.

Fourier transform infrared spectroscopy (FT-IR) was usually employed as an additional probe to evidence the presence of OH groups as well as other organic and inorganic species. The fabricated CeO<sub>2</sub> nanostructured materials were characterized by the spectroscopic techniques in the range 4000–400 cm<sup>-1</sup>. The FT-IR spectra of as-synthesized CeO<sub>2</sub> nanostructures and CTAB are shown in Fig. 2. Some bands are observed in the region 2800–3020 cm<sup>-1</sup>, and are attributed to CTAB surfactant.<sup>62,63</sup> The asymmetric (2918.7 cm<sup>-1</sup>) and symmetric (2846.4 cm<sup>-1</sup>) stretching vibrations of C–CH<sub>2</sub> and C–CH<sub>3</sub> asymmetric stretching and N–CH<sub>3</sub> symmetric stretching vibrations (3011.6 cm<sup>-1</sup>) are assigned to the solid surfactant CTAB. The sharp bands in the region of 1450–1500 cm<sup>-1</sup> are attributed to the deformation of –CH<sub>2</sub>– and –CH<sub>3</sub><sup>63</sup> of the incorporated surfactants. Some bands attributed to the CTAB surfactant are not observed in the region 2800–3020 cm<sup>-1</sup> from Fig. 2(b), 2(c), and 2(d) for the samples of as-synthesized CeO<sub>2</sub> nanostructures, respectively. The absence of

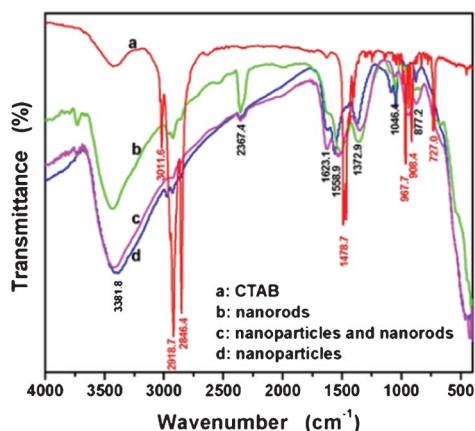


Fig. 2 (a)–(d) FTIR spectra of the as-synthesized  $\text{CeO}_2$  nanostructures.

$\text{CH}_2$  vibrations at  $2846.4\text{ cm}^{-1}$  and  $2918.7\text{ cm}^{-1}$  indicates that the surfactant is not present in the as-synthesized samples.<sup>64</sup> The bands at  $3381.8\text{ cm}^{-1}$  and  $1623.1\text{ cm}^{-1}$  can be attributed to the O–H vibration of adsorbed water on the sample surface.<sup>65</sup> In addition to the bands in the  $850$ – $1600$  and  $2800$ – $3000\text{ cm}^{-1}$  regions, the band due to the stretching frequency of Ce–O can be seen below  $450\text{ cm}^{-1}$ , which confirms the formation of  $\text{CeO}_2$ . The peak positions and peak shapes of the FT-IR spectra are consistent with the standard spectrum of  $\text{CeO}_2$  Sadler (SDBS 40343). The FT-IR absorption bands at about  $1558$ ,  $1373$ ,  $1046$ , and  $877\text{ cm}^{-1}$ , are similar to those of commercial  $\text{CeO}_2$  powders<sup>66</sup> and  $\text{CeO}_2$  nanoparticles.<sup>67</sup>

The morphology of the  $\text{CeO}_2$  nanostructured products were characterized by scanning electron microscopy (SEM). SEM images of the examined samples are shown in Fig. S1, ESI†. As shown in Fig. S1(a), ESI†, the SEM image revealed various particle sizes of the as-prepared sample. The large particles are composed of small crystallites and show particle aggregates of irregular shapes and large sizes ( $1$ – $6\text{ }\mu\text{m}$ ). Fig. S1(b), ESI† gives the image of another  $\text{CeO}_2$  sample, in which nanorods are seen to be up to  $12\text{ nm}$  in diameter and  $200\text{ nm}$  in length. Fig. S1(c), ESI† shows two morphologies of the particles with irregular shapes and the nanorods with lengths of  $100$ – $150\text{ nm}$ , which indicates that the as-synthesized  $\text{CeO}_2$  nanostructure is a mixture of nanoparticles and nanorods.

The morphology and the structure of the final products were further examined with transmission electron microscopy (TEM) and high-resolution TEM (HRTEM). The TEM overview image in Fig. 3(a) displays nanoparticles that are uniform in morphology and narrow in size distribution. To obtain further insight into the local crystallinity of the  $\text{CeO}_2$  nanoparticles, a high-resolution image was recorded. The HRTEM micrograph shows well-developed lattice fringes, which are randomly oriented with respect to each other (Fig. 3(b)). Fig. 3(c) and 3(d) show the typical TEM and HRTEM images of the  $\text{CeO}_2$  nanorods. TEM observation revealed that they all exhibit one-dimensional rod-like nanostructures with poor dispersion. The synthesized  $\text{CeO}_2$  nanostructures display the uniform morphology of nanorods with  $10$ – $12\text{ nm}$  in width and  $100$ – $200\text{ nm}$  in length. That is to say, the aspect ratio of these nanorods is about  $8$ – $15$ . more detailed HRTEM image of  $\text{CeO}_2$  nanorods is displayed in Fig. S2, ESI†. It reveals individual nanorods with

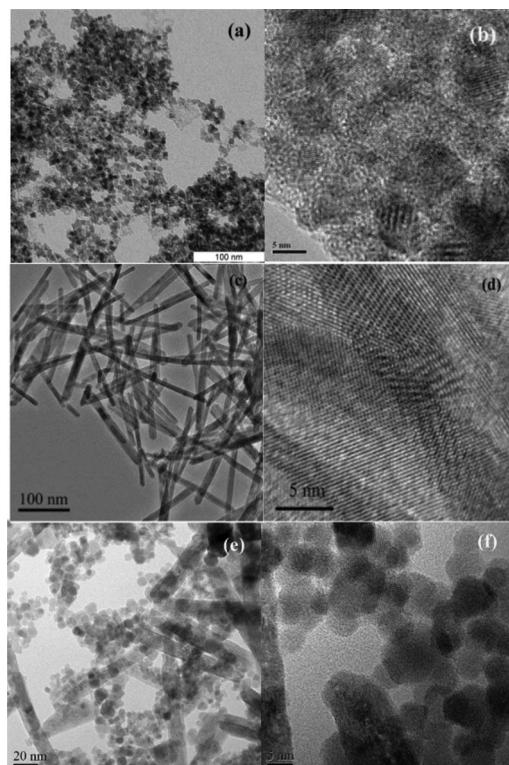


Fig. 3 TEM and HRTEM images of the nanostructured  $\text{CeO}_2$  nanoparticles (a–b), nanorods (c–d), and mixture thereof (e–f).

well-developed lattice fringes, whose spacings are consistent with those of bulk  $\text{CeO}_2$ . The analyzed single nanorod from the upper right corner in Fig. S2, ESI† is recorded with the electron beam parallel to the  $[-111]$  axis, deduced from the lattice fringes and the corresponding fast fourier transform (FFT) pattern. Moreover, on the basis of the identified spots in the FFT pattern, the growth direction of the nanorod is found to be  $[1-12]$  direction. The morphology of the  $\text{CeO}_2$  nanoparticles and nanorods is displayed in Fig. 3(e) and 3(f), where the nanoparticles have well-distributed size and good dispersion, and the nanorods are  $12$ – $14\text{ nm}$  in width and  $100$ – $150\text{ nm}$  in length and have an aspect ratio of  $7$ – $12$ .

The experimental synthetic conditions and the final morphologies of the as-synthesized  $\text{CeO}_2$  nanostructures are presented in Table 1. It is found that the  $\text{CTAB}/\text{Ce}^{3+}$  value and the aging time have a great influence on the morphologies of the final products.

Template-based systems are frequently used to control nucleation and growth of inorganic particles. In this approach, the template simply serves as a scaffold in which different materials are generated *in situ* and are shaped into a nanostructure with its morphology complementary to that of the template. When the solutions (CTAB and  $\text{NH}_3\cdot\text{H}_2\text{O}$  mixed solution, and  $\text{CeCl}_3\cdot6\text{H}_2\text{O}$  solution) were mixed, the precipitate appeared instantly and gradually increased with the addition of the  $\text{CeCl}_3\cdot6\text{H}_2\text{O}$  solution, which indicated that the reaction occurred. Under the action of the modulation preparation, cerium salts and precursors form  $\text{CeO}_2$  nanoparticles through nucleation and growth in a relatively short time. In the process of aging, the nanoparticles stack with each other and grow



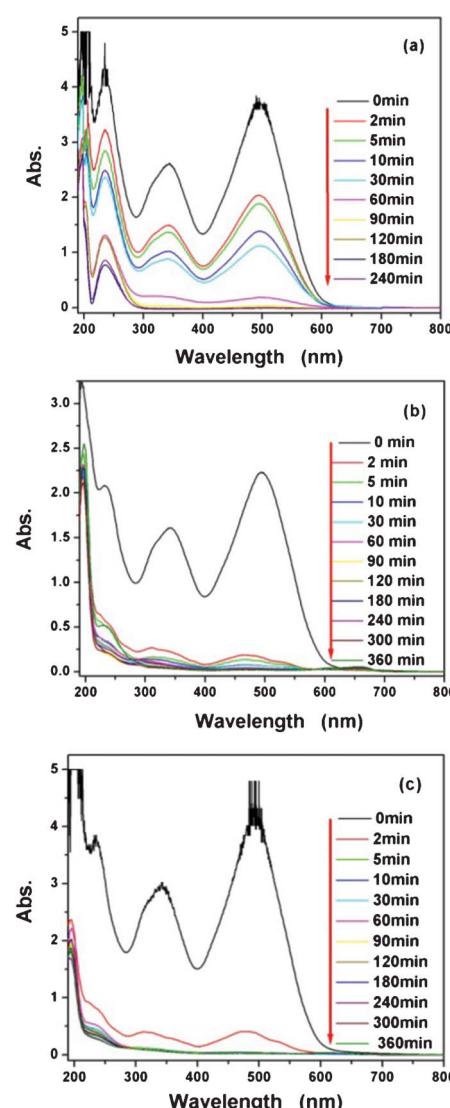
**Fig. 4** Schematic diagram of the proposed mechanism for the formation of the  $\text{CeO}_2$  nanostructures at room temperature.

directionally. For the lower ratio of CTAB and  $\text{Ce}^{3+}$  ( $\text{CTAB} : \text{Ce}^{3+} < 1.5 : 1.0$ ), two morphologies of  $\text{CeO}_2$  nanoparticles and nanorods can be formed; while the higher ratio of CTAB and  $\text{Ce}^{3+}$  ( $\text{CTAB} : \text{Ce}^{3+} \geq 15 : 1$ ),  $\text{CeO}_2$  nanorods can be obtained. A similar finding was reported elsewhere.<sup>68</sup> The overall assembly behaviors of the  $\text{CeO}_2$  nanostructures at room temperature could be illustrated, as in Fig. 4, which reveals the schematic diagram of the formation of  $\text{CeO}_2$  nanostructures. Similar formation mechanisms were reported and the nanostructures in relatively large quantities can be synthesized by templating against rod-like or sphere micelles assembled from the surfactant.<sup>69–71</sup>

**3.1.2 BET surface area.** BET nitrogen adsorption–desorption isotherm measurements were performed to determine the specific surface areas of the as-synthesized  $\text{CeO}_2$  nanostructures. The adsorption–desorption curves with evident hysteresis loops can be found in Fig. S3, ESI†. The adsorbed volume increases dramatically in high relative pressure areas (around 0.9 to 1.0), which is a feature of a textural porosity and indicates the presence of pores within the sample. Gas sorption measurements indicate, at relatively high pressures, the presence of the macropore network of the powders assembly, and, in addition, a pronounced interparticle mesoporosity, as reflected by the hysteresis loop in the intermediary pressure range. The Brunauer–Emmett–Teller (BET) specific surface areas of the as-synthesized  $\text{CeO}_2$  nanostructures are 127.8, 132.9, and 121.1  $\text{m}^2 \text{g}^{-1}$  for nanoparticles, nanorods, and mixture thereof, respectively. This means that the single nanoparticles or nanorods within the spongy microstructure are not really glued together, but are accessible from the outside *via* a secondary mesopore system. In our case, the BET specific surface area of the  $\text{CeO}_2$  nanostructures is much higher than that of the nanorods synthesized by a hydrothermal method ( $50 \text{ m}^2 \text{ g}^{-1}$ ),<sup>72</sup> the porous  $\text{CeO}_2$  nanowire arrays ( $98.1 \text{ m}^2 \text{ g}^{-1}$ ) and nanowires ( $66.9 \text{ m}^2 \text{ g}^{-1}$ ) fabricated by the electrochemical deposition method,<sup>19</sup> the nanotubes ( $80.1 \text{ m}^2 \text{ g}^{-1}$ ), nanorods ( $52.5 \text{ m}^2 \text{ g}^{-1}$ ), and nanoplates ( $37.2 \text{ m}^2 \text{ g}^{-1}$ ).<sup>70</sup> It is believed that photovoltaic responses and catalytic activity performances can be improved by increasing the specific surface area of the catalytic materials. Hence, materials with a nanostructure will provide more surface sites for catalyst–dye contact areas and interface degradation.<sup>19,73,74</sup>

### 3.2 The degradation of CR with different $\text{CeO}_2$ nanostructures

In general, the catalytic activity is tightly related to the structure of the catalyst, such as small size and high surface area.<sup>75</sup> In order to test the catalytic performance of these three cerium samples, the degradation of CR was carried out under visible light illumination at wavelengths longer than 420 nm in an aqueous  $\text{CeO}_2$  dispersion. In order to compare the degradation effects, varying different parameters, such as catalyst amount, initial concentration of CR solution and treatment time, were studied. UV-visible absorbance spectra of the CR solutions, as a function of operation time, are depicted in Fig. 5. As shown in Fig. 5, the results of the degradation experiments under various conditions showed that CR can be effectively removed by the  $\text{CeO}_2$  nanostructures. Rapid removal of CR was observed at the



**Fig. 5** UV-Vis absorption spectra of CR solutions (CRS) before and after treatment with  $\text{CeO}_2$  at different time intervals: (a) CRS ( $100 \text{ mg L}^{-1}$ , 50 mL) in the presence of 250 mg of  $\text{CeO}_2$  nanoparticles, (b) CRS ( $50 \text{ mg L}^{-1}$ , 50 mL) in the presence of 20 mg of  $\text{CeO}_2$  nanorods, and (c) CRS ( $80 \text{ mg L}^{-1}$ , 50 mL) in the presence of 150 mg of the  $\text{CeO}_2$  mixture of nanoparticles and nanorods.

beginning of the reaction time. Following the rapid removal, the removal rate decreased, and an apparent equilibrium was reached after 5–120 min depending on the  $\text{CeO}_2$  nanostructures. When 250 mg of  $\text{CeO}_2$  nanoparticles was added to 50 mL of CR solution, with an initial concentration of  $100 \text{ mg L}^{-1}$ , the  $\text{CeO}_2$  nanoparticles removed about 90% of the CR at room temperature without any additives, as shown in Fig. 5(a). This is a removal capacity of 45 mg CR per mg  $\text{CeO}_2$ . Comparing Fig. 5(a), (b), and (c), after 10 min of reaction, the removal efficiency of 20 mg  $\text{CeO}_2$  nanorods reaches above 97%, which is much higher than that of the  $\text{CeO}_2$  nanoparticles and  $\text{CeO}_2$  mixture of nanoparticles and nanorods. The  $\text{CeO}_2$  nanorods show the highest degradation efficiency and capacity (122 mg CR per g  $\text{CeO}_2$  nanorods) for CR dye in aqueous solution. On the other hand, it is evident that the treatment time has a significant influence on the degradation process, considering that other parameters from Fig. 5 are constant.

The continuous UV-Vis spectra of the centrifuged solution after catalytic reactions at the different intervals were used to record and contrast with that of the initial CR solution, which further clarified the removal performance and mechanism of CR. As seen in Fig. 5, the absorption spectra of the original CR solutions are characterized by one main band in the visible region with its maximum absorption at 493 nm, corresponding to the azo bonds of the CR molecule, which is responsible for the dark red color of the aromatic rings pertaining to the azo groups.<sup>76,77</sup> The other two bands in the ultraviolet region, located at 236 nm and 343 nm, respectively, are attributed to the benzene ring and naphthalene ring structure.<sup>78</sup> The characteristic absorption peaks of CR at 343 nm and 493 nm was chosen as the parameter that was monitored. As shown in Fig. 5, the degradation rate ( $C/C_0$ ) increased with respect to the treatment time, and correspondingly, the amplitude of the peak decreased with respect to time. A continuous decrease in the absorbance peaks at 493 nm and 343 nm in Fig. 5 suggests that the azo bonds and the naphthyl rings are destroyed to form phenyl derivatives. From Fig. 5, one can see that the main absorption peaks (343 nm and 493 nm) of the dye in the UV-Vis region have basically disappeared after 10–60 min. UV light illumination of the aqueous CR solution in the presence of the as-synthesized  $\text{CeO}_2$  nanostructures causes the absorption bands of the CR dye in the visible region to decrease with time and finally to disappear, indicating the destruction of its chromophoric structure in the vicinity of the azo-linkage.

That is, the dye is surely degraded by the catalytic action. Furthermore, the decay of the absorbance at 236 nm can be considered as evidence of aromatic fragment degradation in the dye molecule and its intermediates.<sup>55</sup>

The color of the CR aqueous solutions was bleached when the  $\text{CeO}_2$  nanostructures were used to degrade the CRS and the absorption of the solution at 493 nm was close to zero. Thus, the absorption of the CR aqueous solution at 493 nm could represent the degradation degree of CR. As it can be observed, the disappearance of the visible band is accompanied by total discoloration of the solutions from Fig. 6, which also displays the degradation process of CR. It is clearly seen that the cardinal red color of the starting solution gradually disappears along with the increase in the degradation time. The color of the CR completely disappeared after 5 min when using the  $\text{CeO}_2$



**Fig. 6** Photographs of CR solutions (CRS) before and after treatment with  $\text{CeO}_2$  at different time intervals: (a) CRS ( $100 \text{ mg L}^{-1}$ , 50 mL) in the presence of 250 mg of  $\text{CeO}_2$  nanoparticles, (b) CRS ( $50 \text{ mg L}^{-1}$ , 50 mL) in the presence of 20 mg of  $\text{CeO}_2$  nanorods, and (c) CRS ( $80 \text{ mg L}^{-1}$ , 50 mL) in the presence of 150 mg of the  $\text{CeO}_2$  mixture of nanoparticles and nanorods.

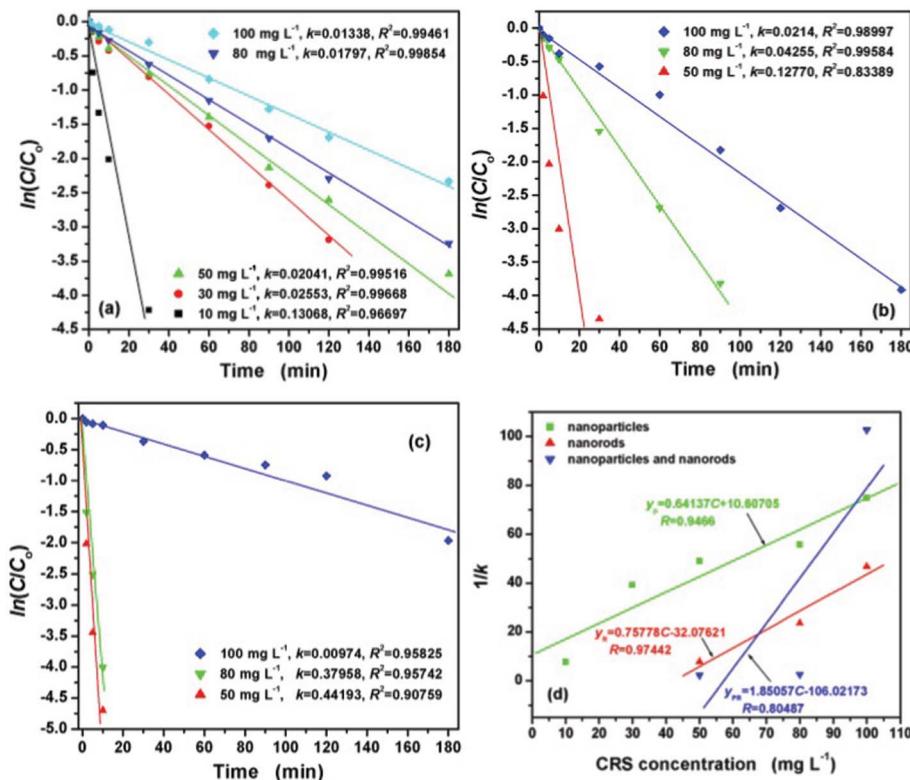
nano rods, while when using the  $\text{CeO}_2$  nanoparticles this took 90 min, suggesting that the  $\text{CeO}_2$  nanorods have the better performance compared to  $\text{CeO}_2$  nanoparticles and the  $\text{CeO}_2$  mixture of nanoparticles and nanorods. The decay of the CR solutions color may be compared with the degradation performance shown in Fig. 5.

### 3.3 The effect of initial CRS concentration on the degradation

Heterogeneous degradation is a complicated process, involving the function of many factors and even their mutual effect. To simplify the process, only the kinetic analysis under the different CRS initial concentrations and catalyst amounts were studied in this work. The kinetics of the aqueous degradation process plays an important role in assessing the efficiency and feasibility of treating dye contaminated wastewater. The kinetics of CR degradation reactions on the catalyst with typical photolytic processes has been described with models based on chemical and photochemical reactions,<sup>79</sup> which can be expressed by the Langmuir–Hinshelwood (L–H) model.<sup>80,81</sup> For the lower initial concentration of CR dye, the reaction rate can be expressed as:<sup>82</sup>

$$\ln(C/C_0) = -kt + B \quad (1)$$

where  $B$  is a constant,  $k$  is degradation rate constant ( $\text{min}^{-1}$ ), and  $t$  is the reaction time. A plot of  $\ln(C/C_0)$  versus  $t$  will yield a slope of  $-k$ . The effect of the initial CRS concentration on the degradation kinetics was studied by varying the concentrations from 10 to  $100 \text{ mg L}^{-1}$  in the presence of 100 mg  $\text{CeO}_2$  nanoparticles, 20 mg  $\text{CeO}_2$  nanorods, and 150 mg  $\text{CeO}_2$  mixture of nanoparticles and nanorods under visible light, respectively. Fig. 7 shows the degradation efficiencies of CR under the different initial concentrations versus time catalyzed by the different  $\text{CeO}_2$  nanostructures,



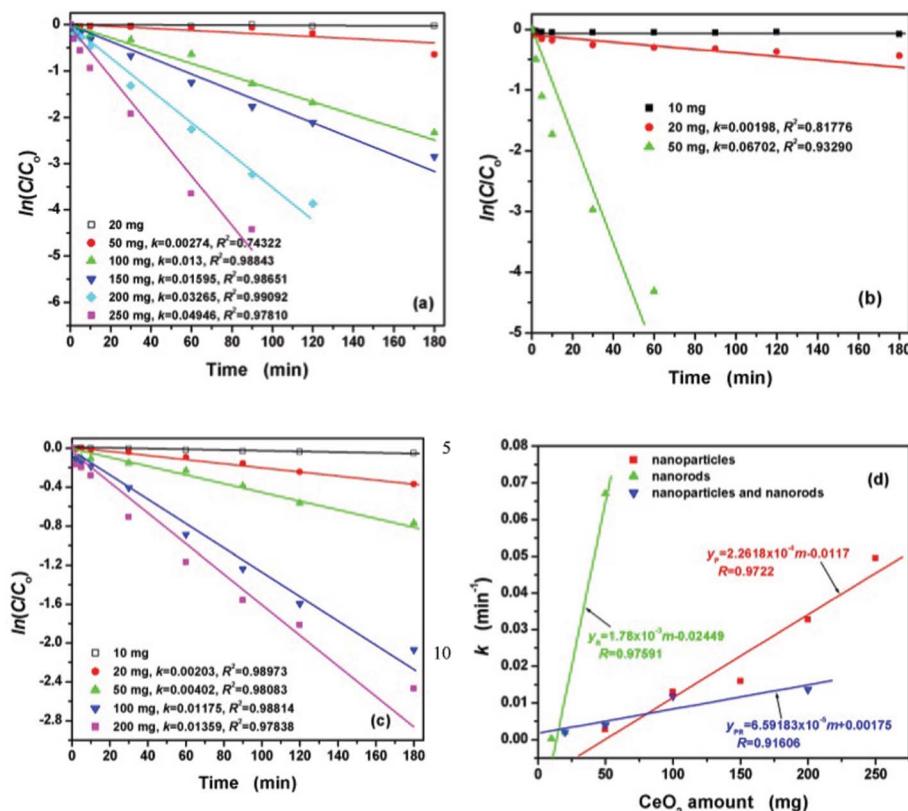
**Fig. 7** The effect of initial CRS concentration on the degradation kinetics for 50 mL CR solution at different time intervals: (a) 100 mg CeO<sub>2</sub> nanoparticles, (b) 20 mg CeO<sub>2</sub> nanorods, (c) 150 mg CeO<sub>2</sub> mixture of nanoparticles and nanorods, and (d) the relationship between the degradation rate constant  $k$  and CRS concentrations from 10 to 100 mg L<sup>-1</sup>.

respectively. They show that the degradation rate depends on the initial concentration of CR. The degradation rate constants at different initial concentrations of CR were determined from the slope of the plots, in accordance to the proposed kinetic model. As seen from Fig. 7, the degradation rate constants decrease with the increase of the initial concentration of CR dye. The results show that the degradation rate depends on the initial concentration of CR solutions, and it is found that CR degradation basically obeys the pseudo-first-order kinetic expression, as can be seen from Fig. 7(a), (b) and (c). The  $k$  values and corresponding regression coefficients (the square of the relative correlative coefficient ( $R^2$ ) of the experimental results) can be found, too. The coefficient values,  $R^2$ , of the experimental runs are more than 0.83, indicating that the degradation of CR dye by the CeO<sub>2</sub> nanostructures satisfactory followed apparent-first-order kinetics. The effect of initial CRS concentration on the degradation kinetics is compared from a plot of  $1/k$  versus different initial dye concentrations in the presence of the different CeO<sub>2</sub> nanostructures as shown in Fig. 7(d). The drastic decrease in the degradation activity with CRS concentration is ascribed to the increase in the local concentration of CR on the CeO<sub>2</sub> surface, leading to the formation of higher order aggregates owing to hydrophobic interactions between the aromatic rings.<sup>57</sup> The degradation effect in the presence of CeO<sub>2</sub> nanorods is better than that of CeO<sub>2</sub> nanoparticles and the CeO<sub>2</sub> mixture of nanoparticles and nanorods throughout all experimental concentration ranges from Fig. 7(d). Perhaps, the following argumentation explains the high degradation ratio for CeO<sub>2</sub> nanorods: the CeO<sub>2</sub> nanorods have the largest surface area and surface-to-volume ratio. It is generally noted that an increase in dye concentration leads to a decrease in the

degradation rate. A possible reason is that the high initial concentration of the dye solution caused more of the degradation reaction to take place to achieve the same level of degradation between the dye molecules and CeO<sub>2</sub> catalyst. However, a higher concentration might also cause excessive adsorption of the CR molecules on the surface of the CeO<sub>2</sub> nanostructures to limit its efficiency. Therefore, as the initial concentration of the CR dye increases, the requirement of the CeO<sub>2</sub> catalyst surface needed for the degradation also increases. It was also found that a longer reaction time is necessary to degrade the higher concentration CRS under the same degradation conditions.

### 3.4 The effect of the CeO<sub>2</sub> nanostructure quantities on the degradation

Further comparative experiments were also carried out to investigate the effect the quantity of CeO<sub>2</sub> nanostructures had on the degradation of the CR dye, which optimized the addition of CeO<sub>2</sub> catalyst used for the highest degradation ratio. The effect of the quantity of CeO<sub>2</sub> nanostructures on the degradation kinetics was studied by varying from 10 to 250 mg in 50 mL CRS, respectively. The corresponding results of the reaction kinetics studies are given in Fig. 8. As shown there, it can be seen that the CeO<sub>2</sub> nanostructures content also plays a very important role in CR removal under visible light irradiation. The degradation rate of CR markedly slowed down due to the small amount of CeO<sub>2</sub> nanostructures. With increased amounts of CeO<sub>2</sub> nanostructures from 10 to 250 mg, the removal ratio of CR increased from 4.4 to 99%. This was due to the increase in



**Fig. 8** The effect of the amount of  $\text{CeO}_2$  nanostructures on the degradation kinetics for 50 mL CRS ( $100 \text{ mg L}^{-1}$ ) at different time intervals: (a)  $\text{CeO}_2$  nanoparticles, (b)  $\text{CeO}_2$  nanorods, (c)  $\text{CeO}_2$  mixture of nanoparticles and nanorods, and (d) the relationship between the degradation rate constant  $k$  and the amounts of  $\text{CeO}_2$  nanostructures from 10 to 250 mg.

catalyst amount, which contributed to the increase in the number of available active sites on the catalyst surface and also the number of the dye molecules degraded. The degradation rate constants  $k$  calculated from the curves referred to 10 or 20 mg  $\text{CeO}_2$  nanostructures are small for all cases, indicating that the CR molecules are difficult to be decomposed under visible light irradiation. Nevertheless, for the catalytic degradation related curves shown in Fig. 8(d), the degradation rate constants reveal that CR molecules can be degraded quickly in the presence of various  $\text{CeO}_2$  nanostructures from 50 to 250 mg. Of course, the  $\text{CeO}_2$  nanorods under visible light irradiation display the highest catalytic activity in these three prepared  $\text{CeO}_2$  nanostructures catalysts.

The degradation rate constant ( $6.702 \times 10^{-2} \text{ min}^{-1}$ ) of 50 mg  $\text{CeO}_2$  nanorods is far beyond those corresponding to the degradation reaction of 50 mg  $\text{CeO}_2$  nanoparticles ( $(2.74 \times 10^{-3} \text{ min}^{-1})$  and 50 mg of the  $\text{CeO}_2$  mixture of nanoparticles and nanorods ( $4.02 \times 10^{-3} \text{ min}^{-1}$ ), respectively, which also belong to first-order reaction kinetics. However, according to the experimental results, the shape of the lines indicates that the pseudo-first-order equation did not fit well to the whole range of the removal process and was generally applicable over the initial stage of the reaction time (less than 30 min).

### 3.5 The degradation of CR using regenerated $\text{CeO}_2$ nanostructures

The performance of  $\text{CeO}_2$  nanostructures after CR degradation is hardly reported. In order to optimize the application of  $\text{CeO}_2$

catalysts for wastewater treatment for an industrial process, the regeneration and reuse of the catalysts are one of the key issues.<sup>83</sup> Based on these aims, an attempt was made in this study to elucidate the differences between the as-synthesized and regenerated  $\text{CeO}_2$  and the consequences on the catalytic activity. The  $\text{CeO}_2$  nanostructures after the catalysis reaction were regenerated by simple drying at  $60^\circ\text{C}$  in air for 12 h, and the regenerated catalysts were used in the degradation of CR dye under similar conditions. The regenerated  $\text{CeO}_2$  nanostructured materials had almost the same degradation performance for CR solutions, as shown in Fig. S4 and Fig. S5, ESI†. The results demonstrate the excellent degradation performance for CR solutions using  $\text{CeO}_2$  nanostructures as catalyst. The excellent degradation performances make it a promising degradation material to remove organic pollutants in wastewater treatments.

### 3.6 The degradation mechanism of CR

To further clarify the removal mechanism of CR, the regenerated  $\text{CeO}_2$  nanoparticles, the as-synthesized  $\text{CeO}_2$  nanoparticles, and the CR dye were characterized by XRD. As shown in Fig. S6, ESI†, all diffraction peaks of the regenerated  $\text{CeO}_2$  nanoparticles can be perfectly indexed to a pure cubic phase, which perfectly indexed to  $\text{CeO}_2$ . Comparing the CR diffraction peaks, no CR peaks are observed from the regenerated  $\text{CeO}_2$  nanoparticles. This indicates that the  $\text{CeO}_2$  catalyst does not adsorb the CR dye after the degradation reaction of CR. For the  $\text{CeO}_2$  nanorods and the  $\text{CeO}_2$  mixture of nanoparticles and nanorods, the same result was obtained.

The FT-IR technique was used to identify specific changes of the functional groups in the CR dye molecule and  $\text{CeO}_2$  nanostructures. The FT-IR spectra of the CR dye, as-synthesized  $\text{CeO}_2$  nanoparticles, and regenerated  $\text{CeO}_2$  nanoparticles were measured and contrasted. From Fig. S7, ESI†, it can be seen that no characteristic peaks of CR dye occur in the FT-IR spectrum of the regenerated  $\text{CeO}_2$  nanoparticles dried after the catalysis reaction, which is also confirmed by the XRD analysis. It indicates that the dye molecules are not adsorbed by the catalyst, implying that the degradation of the dye is caused by the catalytic reaction, not by adsorption. The same is valid for  $\text{CeO}_2$  nanorods and the  $\text{CeO}_2$  mixture of nanoparticles and nanorods. The other two possible removal mechanisms are related to the fact that  $\text{CeO}_2$  is a very good oxidizer, which could remove organic compounds from wastewater through oxidation or photocatalysis decomposition. The main demand for the latter is that the minimum photon energy, in this case visible light with a wavelength longer than 420 nm, required for this process equals the bandgap energy of  $\text{CeO}_2$ , which for our case of  $\text{CeO}_2$  nanoparticles amounts to 3.15 eV, as deduced from the corresponding UV-Vis spectrum (Fig. S8(a), ESI†) by applying the Tauc equation.<sup>84</sup> Accordingly, one obtains a typical plot of  $(\alpha h\nu)^2$  as a function of photon energy, and extrapolates the linear portion of the curve to give the value of the direct bandgap energy (Fig. S8(b), ESI†). It implies that  $\text{CeO}_2$  nanoparticles mainly adsorb light in the UV range, which makes photocatalysis decomposition negligible because our water treatment experiments were performed under a visible light at room temperature. Therefore, the most probable catalytic pathway for the CR removal is oxidative degradation. The main assumption for such a pathway is the presence of redox switch  $\text{Ce}^{3+}/\text{Ce}^{4+}$  cations within the  $\text{CeO}_2$  nanoparticles. In order to corroborate this, samples were analyzed by means of X-ray photoelectron spectroscopy (XPS). The Ce 3d core level peak can be confirmed by the XPS analysis, as shown in Fig. 9. Due to its highly non-stoichiometric nature, both valences (3+ and 4+) are present in  $\text{CeO}_2$ . The main peaks of  $\text{Ce}^{4+}$   $3d_{3/2}$  and  $\text{Ce}^{4+}$   $3d_{5/2}$  are shown at binding energies of 916.6 and 898.3 eV, respectively. Those of  $\text{Ce}^{3+}$   $3d_{3/2}$  and  $\text{Ce}^{3+}$   $3d_{5/2}$  are located at 901.5 and 882.2 eV. Three additional satellite lines SU1 and SU2, which means 'shake-up', are shown at 907.8 eV on the  $\text{Ce}^{3+}$   $3d_{3/2}$  and at 885.8

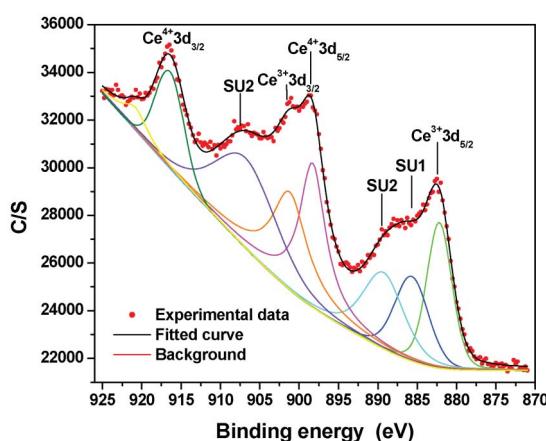


Fig. 9 XPS spectrum of  $\text{CeO}_2$  nanoparticles.

and 889.5 eV on the  $\text{Ce}^{3+}$   $3d_{5/2}$ , respectively. This spectrum is fully consistent with those reported previously.<sup>85–87</sup>

To rationalize XPS findings, one can state that the presence of  $\text{Ce}^{3+}$  is a result of oxygen vacancies and this effect is enhanced in nanoparticles because a larger fraction of the atoms are on the surface as the particle size is reduced and the surface atoms have reduced coordination. This leads to  $\text{CeO}_{2-x}$  as the proper structural description for ceria nanoparticles. The oxygen vacancies lead to the transformation  $\text{Ce}^{4+} \leftrightarrow \text{Ce}^{3+}$ , a key issue in the oxidative degradation of CR. To further validate the proposed mechanism, an additional set of degradation experiments under the exclusion of any visible light (in the dark) were carried out. The representative degradation kinetics, measured under visible light and for comparison in the dark, for 50 mL CR solutions (100 mg  $\text{L}^{-1}$ ) using various  $\text{CeO}_2$  nanostructures are displayed in Fig. S9, ESI†. It reveals that the  $\text{CeO}_2$  nanostructures also exhibit a high degradation activity without any visible light irradiation. Therefore, this comparison experiment additionally confirms the proposed degradation mechanism of CR in the presence of  $\text{CeO}_2$  nanostructures as the oxidative degradation. From Fig. S9, ESI†, one can also notice that the degradation rate without any visible light irradiation is slightly lower than the degradation rate under visible light irradiation. The most probable explanation for this finding is that  $\text{CeO}_2$  nanostructures under visible light conditions might adsorb some UV light and thus they not only degrade CR oxidatively but also photocatalytically.

As shown in Fig. 10, CR and the degradation products formed at the end of the degradation reaction process were analyzed by LC-MS and identified by the interpretation of their mass spectra

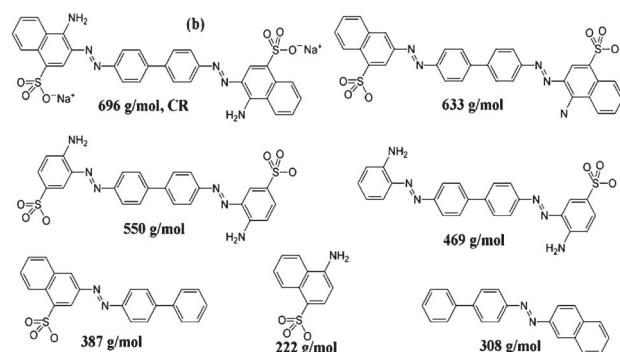
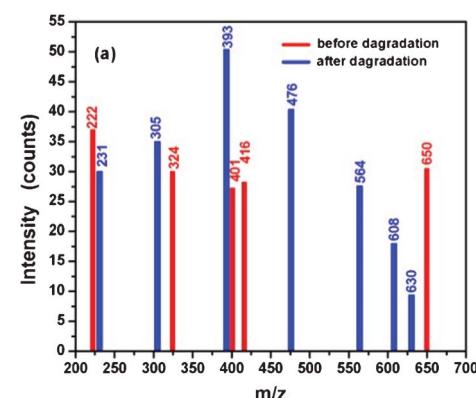
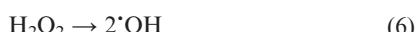
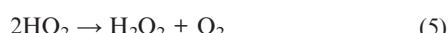
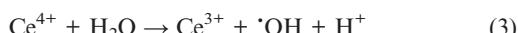


Fig. 10 Spectra of CRS before and after degradation (a) and the CR and possible degradation products of CR (b).

showing their molecular ion peaks with respect to  $m/z$  (where  $m$  is molecular weight of the intermediates in the mass spectra). CR is a sulphonated compound with two sulphonic acid groups. The MS spectrum (Fig. 10(a)) shows two negative pseudo-molecular ions of 650 and 324  $m/z$ . The ion at 650  $m/z$  was due to  $[M - 2Na + H]^-$ , while the ion at 324  $m/z$  corresponds to  $[M - 2Na]^{2-}$ .<sup>88</sup> The results from the MS analysis revealed that seven intermediate by-products were detected in the degradation of the CRS. The spectrum of the CRS is different from that of CRS degraded in the presence of  $\text{CeO}_2$  nanostructures, shown in Fig. 10(a). The difference can be explained by the decomposition of the species at different  $m/z$  ratios in the ion trap and the atmospheric pressure of the chemical ionization source of the instrument, *i.e.*, the presence of  $\text{CeO}_2$  nanostructures changes the decomposition of CR. Li *et al.* reported that CR degradation events include the cleavage of C–S bonds, aromatic ring opening, cleavage of –N=N– double bonds, cleavage of various C–N and C–C bonds in addition to decarboxylation.<sup>89</sup>

Although certified references are not available for the degradation products that occurred in the spectra, the species of the possible degradation products of CR are considered similar to the results reported by Li *et al.*<sup>89</sup> Fig. 10(b) shows the possible species formed after degradation of the CRS in the presence of  $\text{CeO}_2$ , suggesting that CR degrades together with remarkable decolorization and insignificant mineralization to give  $\text{CO}_2$  and  $\text{H}_2\text{O}$ . Similar results can be found in other reports.<sup>89,90</sup>

Based on our experimental results and previous literature records, the degradation mechanism of CR in the presence of  $\text{CeO}_2$  nanostructures can be proposed and is described below:



Rather unstable surface resided  $\text{Ce}^{3+}$  ions in the presence of air (in water solution of CR-eqn (2)) oxidizes to  $\text{Ce}^{4+}$  producing  $\text{O}_2^-$ . Then, the produced  $\text{Ce}^{4+}$  cations can react easily with surface bound  $\text{H}_2\text{O}$  to produce  $\cdot\text{OH}$  radicals (eqn (3)). The superoxide radical anion  $\text{O}_2^-$  can yield hydrogen peroxide  $\text{H}_2\text{O}_2$  (eqn (4) and (5)) in oxygen-equilibrated media. These intermediates can interact to produce a hydroxyl radical  $\cdot\text{OH}$  (eqn (6)). It is well known that the  $\cdot\text{OH}$  radical is a powerful oxidizing agent capable of degrading most pollutants. The CR dye can be oxidized by  $\cdot\text{OH}$  in deep levels (eqn (7)), and then be degraded to form smaller organic molecules, and these organic molecules can eventually be mineralized into  $\text{CO}_2$  and  $\text{H}_2\text{O}$ . The location of these reactive species determines that the degradation of CR only occurs on the surface of  $\text{CeO}_2$ , which also explains why the morphology and BET area have a great influence on their catalytic activity, as discussed previously.

## 4. Conclusions

$\text{CeO}_2$  nanoparticles with various morphologies were successfully synthesized by a surfactant CTAB-assisted coprecipitation process at room temperature. XRD, SEM, TEM, and BET results showed that stable and crystalline phase pure  $\text{CeO}_2$  nanoparticles with a cubic fluorite structure were obtained. The most probable formation mechanism of these  $\text{CeO}_2$  nanoparticles was proposed. This new and simple approach has obvious advantages due to its simplicity, low cost and it being environmentally benign. The degradation performances of the as-synthesized and regenerated  $\text{CeO}_2$  with the different nanostructures towards CR in aqueous solution were studied.

The effects of the amount of  $\text{CeO}_2$  nanostructures, initial CR concentration, and degradation time on the degradation of CR were examined. The removal of CR follows a Langmuir–Hinshelwood first-order kinetic law over the initial stage of the reaction time for the different CR concentrations (10–100 mg L<sup>−1</sup>), by using various amounts of  $\text{CeO}_2$  nanoparticles with various morphologies catalyst. The proposed mechanism for the catalytic removal of CR dyes relies on the presence of surface  $\text{Ce}^{3+}$  cations which *via* a redox  $\text{Ce}^{3+}/\text{Ce}^{4+}$  switch degrade the CR dye. The  $\text{CeO}_2$  nanoparticles are found to be efficient in generating less toxic secondary pollutants, which can be easily eliminated from the system by being subjected further to traditional physical, chemical and biological treatments.

## Acknowledgements

This work was supported by the Department of Science and Technology of Yunnan Province *via* the Key Project for the Science and Technology (Grant No.2011FA001), the Key Project of Chinese Ministry of Education (Grant No. 210206), and National Natural Science Foundation of China (Grant No. 50662006).

## References

- O. S. Polezhaeva, N. V. Yaroshinskaya and V. K. Ivanov, *Russ. J. Inorg. Chem.*, 2007, **52**, 1184–1188.
- B. Murugan and A. V. Ramaswamy, *J. Am. Chem. Soc.*, 2007, **129**, 3062–3063.
- H. Y. Yahiro, K. Baba, H. Eguchi and J. Arai, *J. Electrochem. Soc.*, 1988, **135**, 2077–2080.
- H. Gu and M. D. Soucek, *Chem. Mater.*, 2007, **19**, 1103–1108.
- X. D. Feng, D. C. Sayle, Z. L. Wang, M. S. Paras, B. Santora, A. C. Sutorik, T. X. T. Sayle, Y. Yang, Y. Ding, X. D. Wang and Y. S. Her, *Science*, 2006, **312**, 1504–1508.
- N. Izu, W. Shin, N. Murayama and S. Kanzaki, *Sens. Actuators, B*, 2002, **87**, 95–98.
- S. Tsunekawa, T. Fukuda and A. Kasuya, *J. Appl. Phys.*, 2000, **87**, 1318–1321.
- A. Trovarelli, *Catal. Rev. Sci. Eng.*, 1996, **38**, 439–520.
- J. Kašpar, P. Fornasiero and M. Graziani, *Catal. Today*, 1999, **50**, 285–298.
- M. L. Granados, F. C. Galisteo, P. S. Lambrou, R. Mariscal, J. Sanz, I. Sobrados, J. L. G. Fierro and A. M. Efstratiou, *J. Catal.*, 2006, **239**, 410–421.
- Q. Fu, H. Saltsburg and M. Flytzani-Stephanopoulos, *Science*, 2003, **301**, 935–938.
- E. P. Murray, T. Tsai and S. A. Barnett, *Nature*, 1999, **400**, 649–651.
- A. Corma, P. Atienzar, H. Garcia and J. Y. Chane-Ching, *Nat. Mater.*, 2004, **3**, 394–397.
- A. H. Morshed, M. E. Moussa, S. M. Bedair, R. Leonard, S. X. Liu and N. Elmasry, *Appl. Phys. Lett.*, 1997, **70**, 1647–1649.

15 S. H. Lee, Z. Y. Lu, S. V. Babu and E. Matijevic, *J. Mater. Res.*, 2002, **17**, 2744–2749.

16 S. Yabe and T. Sato, *J. Solid State Chem.*, 2003, **171**, 7–11.

17 S. J. Huang, K. Hara and A. Fukuoka, *Energy Environ. Sci.*, 2009, **2**, 1060–1068.

18 G. A. Umeda, W. C. Chueh, L. Noailles, S. M. Haile and B. S. Dunn, *Energy Environ. Sci.*, 2008, **1**, 484–486.

19 X. H. Lu, D. Z. Zheng, J. Y. Gan, Z. Q. Liu, C. L. Liang, P. Liu and Y. X. Tong, *J. Mater. Chem.*, 2010, **20**, 7118–7122.

20 R. Si, Y. W. Zhang, L. P. You and C. H. Yan, *Angew. Chem., Int. Ed.*, 2005, **44**, 3256–3260.

21 Q. Wu, F. Zhang, P. Xiao, H. S. Tao, X. Z. Wang, Z. Hu and Yinong Lu, *J. Phys. Chem. C*, 2008, **112**, 17076–17080.

22 D. C. Sayle, X. D. Feng, Y. Ding, Z. L. Wang and T. X. T. Sayle, *J. Am. Chem. Soc.*, 2007, **129**, 7924–7935.

23 B. Tang, L. H. Zhuo, J. C. Ge, G. L. Wang, Z. Q. Shi and J. Y. Niu, *Chem. Commun.*, 2005, 3565–3567.

24 M. Yada, S. Sakai, T. Torikai, T. Watari, S. Furuta and H. Katsuki, *Adv. Mater.*, 2004, **16**, 1222–1226.

25 L. Yan, R. B. Yu, J. Chen and X. R. Xing, *Cryst. Growth Des.*, 2008, **8**, 1474–1477.

26 W. Q. Han, L. Wu and Y. Zhu, *J. Am. Chem. Soc.*, 2005, **127**, 12814–12815.

27 Y. Ma, X. D. Wang, S. H. Li, M. S. Toprak, B. Zhu and M. Muhammed, *Adv. Mater.*, 2010, **22**, 1640–1644.

28 P. Kluson, P. Kacer, T. Cajthaml and M. Kalaji, *J. Mater. Chem.*, 2001, **11**, 644–651.

29 T. Masui, K. Fujiwara, K. Machida, G. Adachi, T. Sakata and H. Mori, *Chem. Mater.*, 1997, **9**, 2197–2204.

30 S. G. Dixit, A. R. Mahadeshwar and S. K. Haram, *Colloids Surf., A*, 1998, **133**, 69–75.

31 K. H. Chung and D. C. Park, *Catal. Today*, 1996, **30**, 157–162.

32 G. R. Bamwenda and H. J. Arakawa, *J. Mol. Catal. A: Chem.*, 2000, **161**, 105–113.

33 M. D. Hernández-Alonso, A. B. Hungria, A. Martínez-Arias, M. Fernández-García, J. M. Coronado, J. C. Conesa and J. Soria, *Appl. Catal., B*, 2004, **50**, 167–175.

34 Y. Q. Zhai, S. Y. Zhang and H. Pang, *Mater. Lett.*, 2007, **61**, 1863–1866.

35 P. Borker and A. V. Salker, *Mater. Chem. Phys.*, 2007, **103**, 366–370.

36 L. Yue and X. M. Zhang, *J. Alloys Compd.*, 2009, **475**, 702–705.

37 L. W. Qian, J. Zhua, W. M. Du and X. F. Qian, *Mater. Chem. Phys.*, 2009, **115**, 835–840.

38 J. Kašpar, P. Fornasiero and N. Hickey, *Catal. Today*, 2003, **77**, 419–449.

39 S. W. Boettcher, J. Fan, C. Tsung, Q. Shi and G. D. Stucky, *Acc. Chem. Res.*, 2007, **40**, 784–792.

40 Z. X. Li, L. L. Li, Q. Yuan, W. Feng, J. Xu, L. D. Sun, W. G. Song and C. H. Yan, *J. Phys. Chem. C*, 2008, **112**, 18405–18411.

41 H. Y. Xiao, Z. H. Ai and L. Z. Zhang, *J. Phys. Chem. C*, 2009, **113**, 16625–16630.

42 Y. Zhai, S. Zhang and H. Pang, *Mater. Lett.*, 2007, **61**, 1863–1866.

43 P. Borker and A. V. Salker, *Mater. Chem. Phys.*, 2007, **103**, 366–370.

44 J. B. Fei, Y. Cui, X. H. Yan, Y. Yang, K. W. Wang, Q. He and J. B. Li, *Adv. Mater.*, 2008, **20**, 452–456.

45 Z. J. Yang, J. J. Wei, H. X. Yang, L. Liu, H. Liang and Y. Z. Yang, *Eur. J. Inorg. Chem.*, 2010, 3354–3359.

46 J. H. Sun, Y. K. Wang, R. X. Sun and S. Y. Dong, *Mater. Chem. Phys.*, 2009, **115**, 303–308.

47 V. A. Sakkas, M. A. Islam, C. Stalikas and T. A. Albanis, *J. Hazard. Mater.*, 2010, **175**, 33–44.

48 M. N. Chong, H. Y. Zhu and B. Jin, *Chem. Eng. J.*, 2010, **156**, 278–285.

49 M. T. C. Sansiviero, D. S. dos Santos, A. E. Job and R. F. Aroca, *J. Photochem. Photobiol., A*, 2011, **220**, 20–24.

50 G. Laeraa, B. Jin, H. Zhu and A. Lopez, *Catal. Today*, 2011, **161**, 147–152.

51 A. K. Kondru, P. Kumar and S. Chand, *J. Hazard. Mater.*, 2009, **166**, 342–347.

52 B. Cheng, Y. Le, W. Q. Cai and J. G. Yu, *J. Hazard. Mater.*, 2011, **185**, 889–897.

53 A. Afkhami and R. Moosavi, *J. Hazard. Mater.*, 2010, **174**, 398–403.

54 S. Chatterjee, M. W. Lee and S. H. Woo, *Bioresour. Technol.*, 2010, **101**, 1800–1806.

55 H. Y. Zhu, R. Jiang, L. Xiao, Y. H. Chang, Y. J. Guan, X. D. Li and G. M. Zeng, *J. Hazard. Mater.*, 2009, **169**, 933–940.

56 H. R. Pouretedal and M. H. Keshavarz, *J. Alloys Compd.*, 2010, **501**, 130–135.

57 A. Elazouti, N. Laouedj and B. Ahmed, *J. Chem. Eng. Process Technol.*, 2011, **2**, 1–9.

58 K. Melghit, M. S. Al-Rubaei and I. Al-Amri, *J. Photochem. Photobiol., A*, 2006, **181**, 137–141.

59 A. Mittal, J. Mittal, A. Malviya and V. K. Gupta, *J. Colloid Interface Sci.*, 2009, **340**, 16–26.

60 J. Rodriguez-Carvajal, *Fullprof – program for the Rietveld refinement*, Laboratoire Leon Brillouin, CEA-Saclay, France, 2001.

61 F. Zhang, S. Chan, J. E. Spanier, E. Apak, Q. Jin, R. D. Robinson and I. P. Herman, *Appl. Phys. Lett.*, 2002, **80**, 127–129.

62 K. H. S. Kung and K. F. Hayes, *Langmuir*, 1993, **9**, 263–267.

63 L. J. Bellamy, *The infrared spectra of complex molecules*, Chapman Hall, New York, 1975.

64 Q. Y. Mu and Y. D. Wang, *J. Alloys Compd.*, 2011, **509**, 2060–2065.

65 J. X. Xu, G. S. Li and L. P. Li, *Mater. Res. Bull.*, 2008, **43**, 990–995.

66 D. S. Zhang, H. X. Fu, L. Y. Shi, C. S. Pan, Q. Li, Y. L. Chu and W. Y. Yu, *Inorg. Chem.*, 2007, **46**, 2446–2451.

67 S. Phoka, P. Laokul, E. Swatsitang, V. Promarak, S. Seraphin and S. Maensiri, *Mater. Chem. Phys.*, 2009, **115**, 423–428.

68 A. Vantomme, Z. Y. Yuan, G. Du and B. L. Su, *Langmuir*, 2005, **21**, 1132–1135.

69 D. S. Zhang, H. X. Fu, L. Y. Shi, C. S. Pan, Q. Li, Y. L. Chu and W. J. Yu, *Inorg. Chem.*, 2007, **46**, 2446–2451.

70 C. S. Pan, D. S. Zhang and L. Y. Shi, *J. Solid State Chem.*, 2008, **181**, 1298–1306.

71 Q. Y. Mu, T. Chen and Y. D. Wang, *Nanotechnology*, 2009, **20**, 345602 (7 pages).

72 K. B. Zhou, X. Wang, X. M. Sun, Q. Peng and Y. D. Li, *J. Catal.*, 2005, **229**, 206–212.

73 P. F. Ji, J. L. Zhang, F. Chen and M. Anpo, *J. Phys. Chem. C*, 2008, **112**, 17809–17813.

74 P. X. Huang, F. Wu, B. L. Zhu, X. P. Gao, H. Y. Zhu, T. Y. Yan, W. P. Huang, S. H. Wu and D. Y. Song, *J. Phys. Chem. B*, 2005, **109**, 19169–19174.

75 Y. Wang, Y. Wang, Y. L. Meng, H. M. Ding, Y. K. Shan, X. Zhao and X. Z. Tang, *J. Phys. Chem. C*, 2008, **112**, 6620–6626.

76 J. S. Chang, C. Chou, Y. C. Lin, P. J. Lin, J. Y. Ho and L. H. Tai, *Water Res.*, 2001, **35**, 2841–2850.

77 N. H. Ince, M. I. Stefan and J. R. Bolton, *J. Adv. Oxid. Technol.*, 1997, **2**, 442–448.

78 Y. Q. Cao, Y. Y. Hu, J. Sun and B. Hou, *Bioelectrochemistry*, 2010, **79**, 71–76.

79 B. C. B. Salgado, M. I. C. Nogueira, K. A. Rodrigues, G. M. M. S. Sampaio, H. L. Buarque and R. S. Araújo, *Engenharia Sanitária e Ambiental*, 2009, **14**, 1–9.

80 H. Y. Chen, O. Zahraa, M. Bouchy, F. Thomas and J. Y. Bottero, *J. Photochem. Photobiol., A*, 1995, **85**, 179–186.

81 R. K. Wahli, W. W. Yu, Y. Liu, M. L. Mejia, J. C. Falkner, W. Nolte and V. L. Colvin, *J. Mol. Catal. A: Chem.*, 2005, **242**, 48–56.

82 S. Kaur and V. Singh, *J. Hazard. Mater.*, 2007, **141**, 230–236.

83 S. Mozia, M. Toyoda, M. Inagaki, B. Tryba and A. W. Morawski, *J. Hazard. Mater.*, 2007, **140**, 369–375.

84 J. A. Prins, *Proc. Int. Conf. Physics of Noncrystalline Solids*, Amsterdam, North Holland Pub. Co, New York, 1964.

85 S. Tsunekawa, T. Fukuda and A. Kasuya, *Surf. Sci.*, 2000, **457**, L437–L440.

86 D. R. Mullins, S. H. Overbury and D. R. Huntley, *Surf. Sci.*, 1998, **409**, 307–319.

87 R. Vercaemst, D. Poelman, R. L. Van Meirhaeghe, L. Fiermans, W. H. Laflke and F. Cardon, *J. Lumin.*, 1995, **63**, 19–30.

88 H. J. Ma, M. Wang, R. Y. Yang, W. F. Wang, J. Zhao, Z. Q. Shen and S. D. Yao, *Chemosphere*, 2007, **68**, 1098–1104.

89 D. Li, Y. Guo, C. Hu, C. Jiang and E. Wang, *J. Mol. Catal. A: Chem.*, 2004, **207**, 183–193.

90 S. Erdemoğlu, S. K. Aksu, F. Sayilan, B. İzgi, M. Asiltürk, H. Sayilan, F. Frimmel and S. Gücer, *J. Hazard. Mater.*, 2008, **155**, 469–477.